Research Article

Incorporation of Decanethiol-Passivated Gold Nanoparticles into Cross-Linked Poly(Dimethylsiloxane) Films

Motohiro Tagaya¹ and Masaru Nakagawa²

¹ Department of Metallurgy and Ceramics Science, Tokyo Institute of Technology, 2-12-1-S7-6 Ookayama, Meguro-ku, Tokyo 152-8550, Japan
² Institute of Multidisciplinary Research for Advanced Materials, Tohoku University, 2-1-1 Katahira, Aoba-ku, Sendai 980-8577, Japan

Correspondence should be addressed to Motohiro Tagaya, tagaya.m.aa@m.titech.ac.jp

Received 18 May 2010; Accepted 11 November 2010

Copyright © 2011 M. Tagaya and M. Nakagawa. This is an open access article distributed under the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited.

Cross-linking degree of a poly(dimethylsiloxane) (PDMS) film was controlled, and the incorporation of hydrophobic decanethiol-passivated gold (Au) nanoparticles into the film was investigated. FT-IR spectra indicated that the hydrosilylation reaction between a vinyl group and a hydrosilyl group occurred with the cross-linking. The swelling degree of the film in toluene changed with a cross-linker concentration, indicating the control of the cross-linking degree of PDMS film. By EDX analysis, the amount of incorporated Au nanoparticles increased with decreasing a cross-linker concentration, indicating the enlarged free volume of the film. The Au nanoparticle-PDMS composite film containing a cross-linker at 6 wt% showed brown color attributed to plasmon resonance of Au nanoparticles, suggesting the Au nanoparticles in the film at monodispersion state. The UV-visible absorbance of the composite film decreased without spectralshift by swelling with toluene, and the changes were reversible. The aggregation among Au nanoparticles in the composite film after calcination also depended on the cross-linking degree. Thus, the control of cross-linking degree of PDMS film successfully leaded to a simple way of fabricating the Au nanoparticle-PDMS composite film at the mono-dispersion state.

1. Introduction

Metal nanoparticles are expected to have enhanced nonlinear optical properties due to plasmon resonance [1, 2], and the resonance is important in the development of optical devices [3, 4]. The production of a monodispersed nanoparticle-matrix system with a specific cluster size would have a great impact on various technological fields such as electronics and catalysis because of the electronic structure [5]. It is known that the conduction band present in a bulk metal will be absent in a nanoparticle, and instead, there would be discrete states at the band edge. The formation of metal nanoparticles in a solid support matrix by different techniques such as impregnation, evaporation, traditional melt quenching, ion implantation, ion exchange, and the sol-gel methods has been investigated [6, 7]. Development of easy formation process of the nanoparticle-matrix composite at a low temperature is required.

As a nanoparticle support at a mono-dispersion, inorganic porous materials have been unique potential advantages [8, 9]. The nanoparticles are formed into the support surface by reduction of metal ion in the pore and the sintering. Although the inorganic porous supports are rigid and durable, it is difficult to prepare nonbreakable self-standing composite film. To be applicable for optics or electronics in next generation, the support materials should have the properties of (i) flexibility, (ii) transparency in visible region and (iii) controllable free volume to accommodate metal nanoparticles.

To meet these needs, poly(dimethylsiloxane) (PDMS) is one of typical elastomer materials with the desirable properties such as flexibility, transparency, chemically inert and thermally stability. The PDMS is attractive material for the development of microcontact printing, filtration film, and microfluid [10–15]. We have reported that locally photo-oxidized patterns on a flat PDMS film allowed us to
transcribe a pattern of Au nanoparticles into a solid substrate [16]. The PDMS is prone to swelling in the presence of alkane and aromatic solvents because of the hydrophobic and nonpolar nature of the surface. The use of the cross-linked PDMS is the key techniques for achievement of efficient and easy incorporation of metal nanoparticles on a material surface. The cross-linking degree of the silicone network would be a very important parameter to control the free volume to accommodate the metal nanoparticles.

In this study, we demonstrated the control of cross-linking degree of PDMS film by hydrosilylation reaction and the subsequent direct incorporation of hydrophobic Au nanoparticles into the swelled PDMS film. The incorporated amount and state of the Au nanoparticles in the PDMS film with the cross-linking degree was investigated.

2. Experiment

2.1. Preparation of Cross-Linked PDMS Films. A cross-linked PDMS was prepared by the mixture of liquid PDMS (Sylgard 184A, Dow Corning Co. Ltd.) and cross-linker (Sylgard 184B, Dow Corning Co. Ltd.) in 1.5, 2.0, 3.0, 4.0, 6.0, 8.0, and 10.0 wt% of cross-linker to liquid PDMS. The mixtures were spin-coated (1000 rpm, 20 s) or casted by a weight of 0.15 g for 12 h for toluene for 30 min were also measured.

The cross-linking degree is defined as a decrease rate of hydrosilyl groups before and after the curing, and the degree of the thin films was measured by FT-IR spectral changes. The relative characteristic absorbance of stretching vibration in Si–H at 2161 cm\(^{-1}\) to C–H asymmetric stretching vibration in –CH\(_3\) at 2965 cm\(^{-1}\), which is abbreviated as \(A_{\text{Si–H}}/A_{\text{C–H}}\), was used. The absorbance of composite PDMS film containing a cross-linker of \(x\) wt% was abbreviated as PDMSx. The thickness of the spin-coated thin film and thick film were respectively 450 nm and 1 mm. The cross-linked PDMS films were immersed in toluene for 1 h and used after removal of nonreactive crudes.

The cross-linking degree was defined as a decrease rate of hydrosilyl groups before and after the curing, and the degree of the thin films was measured by FT-IR spectral changes. The relative characteristic absorbance of stretching vibration in Si–H at 2161 cm\(^{-1}\) to C–H asymmetric stretching vibration in –CH\(_3\) at 2965 cm\(^{-1}\), which is abbreviated as \(A_{\text{Si–H}}/A_{\text{C–H}}\), was used. The absorbance of composite PDMS film containing a cross-linker of \(x\) wt% was abbreviated as PDMSx. The thickness of the spin-coated thin film and thick film were respectively 450 nm and 1 mm. The cross-linked PDMS films were immersed in toluene for 1 h and used after removal of nonreactive crudes.

After the immersion, the films were immersed in toluene for 30 min and completely dried by nitrogen blow. The incorporation amount of the Au nanoparticles was estimated by relative weight ratio of Au to Si element by using a Hitachi S-3000N SEM equipped with EDX mapping function, and the peak ratio of attributed to M shell in Au to L shell in Si spectrum on the cross-section was calculated. The Au nanoparticle–PDMS composite films were calcinated at 550 °C for 4 h. UV-visible absorbance of the Au nanoparticle–PDMS composite films, and a toluene solution containing Au nanoparticles of 2.5 \(\times\) \(10^{-3}\) wt% was measured by UV-Visible absorption spectrophotometer (Shimazu UV-3100). The absorbance of composite PDMS film containing the cross-linker concentration at 6 wt% before and after swelling in toluene for 30 min were also measured.

2.2. Incorporation of Hydrophobic Gold Nanoparticles into the Cross-Linked PDMS Films. Decanethiol-(DT)-passivated gold (Au) nanoparticles having an average metallic core diameter of 2.2 nm (Tanaka Kikinzoku Group Co., Ltd.) were used. The thick cross-linked PDMS films were used for the incorporation of the hydrophobic Au nanoparticles, and the films were immersed in toluene solution containing Au nanoparticles at the concentration of 5 wt% for 270 min.

3. Results and Discussion

3.1. Preparation of Cross-Linked PDMS Films. Figure 1(a) shows FT-IR spectrum of PDMS film before the thermal curing. The absorbance spectrum shows the bands of Si–C stretching vibration at 795–800 cm\(^{-1}\), Si–O stretching vibration in Si–OH at 850 and 870 cm\(^{-1}\), Si–H rocking vibration at 192 cm\(^{-1}\), Si–O–Si asymmetric stretching vibration at 1025–1080 cm\(^{-1}\), C–H rocking vibration in –CH\(_3\) at 1260 cm\(^{-1}\), Si–H stretching vibration at 2161 cm\(^{-1}\), and asymmetric stretching vibration in –CH\(_2\) at 2965 cm\(^{-1}\). These bands are attributed to the liquid PDMS contains siloxane base oligomers with plural vinyl group (CH\(_2\)=CH\(_2\)) and cross-linker contains siloxane base polymers with hydrosilyl (Si–H) groups.

Figure 1(b) shows FT-IR spectra of the PDMS films containing 10 wt% of cross-linker to liquid PDMS before and after the curing (65°C for 12 h). The characteristic bands of Si–H stretching vibration at 2161 cm\(^{-1}\) and asymmetric –CH\(_3\) stretching vibration at 2965 cm\(^{-1}\) in Si–CH\(_3\) were clearly observed as compared with the bands before the curing. These bands show side chains of PDMS bone structure. The thermal cure caused a noticeable decrease in the absorbance attributed to Si–H group. The spectral decrease indicated that a hydrosilyl group reacted with a vinyl group to form a –Si–CH\(_2\)=CH\(_2\)– linkage. It has been reported that a platinum-type catalyst in the regent carried out a hydrosilylation reaction to form the cross-link of –Si–CH\(_2\)=CH\(_2\)=CH\(_2\)– linkage [17, 18]. By the thermal curing, the cross-linking with a –Si–CH\(_3\)=CH\(_2\)=CH\(_2\)– linkage occurred.

Figure 1(c) shows cross-linking degree (\((R_{\text{before}} - R_{\text{after}})/ R_{\text{before}}\)) of the PDMS films as a function of a cross-linker concentration to liquid PDMS. The cross-linking degree increased with the added amount of cross-linker. The degrees of cross-linker concentration at 1.5 wt% and 10 wt% were 0.1 and 0.6. These results indicated that the free volume decreased with the –Si–CH\(_2\)=CH\(_2\)=CH\(_2\)– linkage with increasing the cross-linker concentration.

Figure 2 shows relationship between cross-linker concentration to liquid PDMS and swelling degree by toluene. The swelling degree of PDMS2.0 was 600%, and the value was much larger than that of PDMS10.0 (200%). The incorporated amount of toluene in the PDMS film at the
Figure 1: FT-IR spectra of (a) PDMS film before cross-linking and (b) PDMS films (i) before and (ii) after the heat at 65°C for 12 h. (c): cross-linking degree of the PDMS films as a function of a cross-linker concentration to liquid PDMS.
Figure 3 shows (a) UV-visible absorption spectrum of toluene solution containing Au nanoparticles [19]. The siloxane skeletons of PDMS surface have polar properties, and polar molecules are likely to be captured on the PDMS surface by electrostatic interaction. In this study, the surrounding thiol molecules of the Au nanoparticles have a permanent dipole moment, and the dipole interaction would anchor the Au nanoparticles on the PDMS surface in low dielectric constant toluene.

Figure 4 shows UV-visible absorption spectra and photographs of the Au nanoparticle-PDMS$	imes$0 composite film (a) before and (b) after swelling. The dashed line shows that UV-visible absorption spectrum of toluene solution containing Au nanoparticles of $2.5 \times 10^{-3}$ wt%. The spectrum of composite film before swelling showed a broad absorption due to plasmon resonance band at around 513 nm superimposed on that of transitions from 5d to 6s bands [20]. The spectrum is similar to that of the Au nanoparticle solution, suggesting that Au nanoparticles are mono-dispersed in the PDMS film. The absorbance decreased by ca. 50% without spectral shift by the swelling, and the spectral change was reversible with swelling/shrinkage. The absorbance due to plasmon resonance band found to respond to toluene.

Figure 5 shows change of weight ratio of Au to Si element in the composite film as a function of a cross-linker concentration to liquid PDMS. The EDX mappings images of Au element in the composite films showed the uniform incorporation (data not shown). The weight ratio of the Au to Si in the cross-section decreased with increasing cross-linker concentration. The incorporation amount of Au nanoparticles was successfully controlled by the cross-linking degree.

Figure 6 shows diffuse reflectance UV-visible absorption spectra of the composite films containing (a) 2–10 wt% of a cross-linker to liquid PDMS after the calcination. The composite film was changed to a solid state by the calcination, indicating new formation of siloxane bonds with decomposition of methyl groups in methyldisiloxane. The PDMS$	imes$0 and PDMS$	imes$4.0 showed the absorbance at longer region, indicating the aggregation of Au nanoparticles by the calcination. On the other hand, the absorption spectra of cross-linker concentration more than 6 wt% clearly showed the plasmon resonance band at 530 nm, indicating that the mono-dispersed Au nanoparticles remains. The cross-linking in the polymer network would suppress the aggregation among Au nanoparticles against microenvironment changes by the calcination.

### 4. Conclusion

We demonstrated that the cross-linking degree of PDMS film was controlled by a cross-linker concentration. The hydrophobic Au nanoparticles were efficiently and easily incorporated into the PDMS film and were strongly adhered on the PDMS surface. The incorporated amount and state of the Au nanoparticles in the PDMS film were controlled by the cross-linking degree. The Au composite film containing 6 wt% of cross-linker showed the plasmon resonance of Au nanoparticles, suggesting the mono-dispersion of Au nanoparticles in the film. The aggregation state of Au nanoparticles in the composite film before and after calcination clearly depended on the cross-linking degree. To our
Figure 3: (a) UV-visible absorption spectra and (b) photographs of the composite films containing (i) 2, (ii) 4, (iii) 6, (iv) 8 and (v) 10 wt% of a cross-linker to liquid PDMS.

Figure 4: UV-Visible absorption spectra and photographs of the Au nanoparticle-PDMS6.0 composite film (a) before and (b) after swelling. The dashed line shows UV-Visible absorption spectrum of toluene solution containing Au nanoparticles at the concentration of $2.5 \times 10^{-3}$ wt%.
6 Smart Materials Research

Weight ratio: Au to Si

0 1/2 1 1.5 2
0.5 1 1.5 2

Cross-linker to liquid PDMS (wt%)

Figure 5: Change of the weight ratio of Au to Si element in the composite films as a function of the cross-linker concentration.

Figure 6: Diffuse reflectance UV-visible absorption spectra of the composite films after the calcination containing (a) 2, (b) 4, (c) 6, (d) 8 and (e) 10 wt% of a cross-linker to liquid PDMS.

knowledge, this article is the first report on incorporation of hydrophobic Au nanoparticles into a PDMS to lead to a simple way of fabricating Au nanoparticle-PDMS composite film.

References

[1] S. Link and M. A. El-Sayed, “Spectral properties and relaxation dynamics of surface plasmon electronic oscillations in gold and silver nanodots and nanorods,” *Journal of Physical Chemistry B*, vol. 103, no. 40, pp. 8410–8426, 1999.

[2] U. Schürmann, W. Hartung, H. Tako, V. Zaporetnchenko, and F. Faupel, “Controlled syntheses of Ag-polytetrafluoroethylene nanocomposite thin films by co-sputtering from two magnetron sources,” *Nanotechnology*, vol. 16, no. 8, pp. 1078–1082, 2005.

[3] Y. Kondo, Y. Kuroiwa, N. Sugimoto et al., “Third-order optical non-linearities of CuCl-doped glasses in a near resonance region,” *Journal of Non-Crystalline Solids*, vol. 196, pp. 90–94, 1996.

[4] B. Kutsch, O. Lyon, M. Schmitt, M. Mennig, and H. Schmidt, “Investigations of the electronic structure of nanoscaled gold colloids in sol-gel coatings,” *Journal of Non-Crystalline Solids*, vol. 217, no. 2-3, pp. 143–154, 1997.

[5] A. Vaskelis, A. Jagminiene, L. Tamasiukaita-Tamasiunaita, and R. Juškenas, “Silver nanostructured catalyst for modification of dielectrics surface,” *Electrochimica Acta*, vol. 50, no. 23, pp. 4586–4591, 2005.

[6] M. Fukushima, H. Yanagi, S. Hayashi, N. Suganuma, and Y. Taniguchi, “Fabrication of gold nanoparticles and their influence on optical properties of dye-doped sol-gel films,” *Thin Solid Films*, vol. 438-439, pp. 39–43, 2003.

[7] R. Trbojevich, N. Pellegrin, A. Frattini, O. De Sanctis, P. J. Morais, and R. M. Almeida, “Preparation and isolation of gold nanoparticles coated with a stabilizer and sol-gel compatible agent,” *Journal of Materials Research*, vol. 17, no. 8, pp. 1973–1980, 2002.

[8] U. Junges, W. Jacobs, I. Voigt-Martin, B. Krutzsch, and F. Schüth, “MCM-41 as a support for small platinum particles; a catalyst for low-temperature carbon monoxide oxidation,” *Journal of the Chemical Society, Chemical Communications*, no. 22, pp. 2283–2284, 1995.

[9] R. Leon, D. Margolese, G. Stucky, and P. M. Petroff, “Nanocrystalline Ge filaments in the pores of a mesosilicate,” *Physical Review B*, vol. 52, no. 4, pp. R2285–R2288, 1995.

[10] A. Kumar and G. M. Whitesides, “Features of gold having micrometer to centimeter dimensions can be formed through a combination of stamping with an elastomeric stamp and an alkane-thiol ‘ink’ followed by chemical etching,” *Applied Physics Letters*, vol. 63, no. 14, pp. 2002–2004, 1993.

[11] N. Stafie, D. F. Stamatialis, and M. Wessling, “Insight into the transport of hexane-solute systems through tailor-made composite membranes,” *Journal of Membrane Science*, vol. 228, no. 1, pp. 103–116, 2004.

[12] S. Mandal and V. G. Pangarkar, “Separation of methanol-benzene and methanol-toluene mixtures by pervaporation: effects of thermodynamics and structural phenomenon,” *Journal of Membrane Science*, vol. 201, no. 1-2, pp. 175–190, 2002.

[13] L. Gales, A. Mendes, and C. Costa, “Removal of acetone, ethyl acetate and ethanol vapors from air using a hollow fiber PDMS membrane module,” *Journal of Membrane Science*, vol. 197, no. 1-2, pp. 211–222, 2002.

[14] M. A. Unger, H. P. Chou, T. Thorsen, A. Scherer, and S. R. Quake, “Monolithic microfabricated valves and pumps by multilayer soft lithography,” *Science*, vol. 288, no. 5463, pp. 113–116, 2000.

[15] J. C. McDonald and G. M. Whitesides, “Poly(dimethyldisiloxane) as a material for fabricating microfluidic devices,”
[16] M. Tagaya, M. Komura, T. Iyoda, and M. Nakagawa, “Printing of Au nanoparticles by Vuv-exposed patterned surfaces of a poly(dimethylsiloxane) film,” Transactions of the Materials Research Society of Japan, vol. 31, pp. 273–276, 2006.

[17] D. J. Campbell, K. J. Beckman, C. E. Calderon et al., “Replication and compression of surface structures with polydimethylsiloxane elastomer,” Journal of Chemical Education, vol. 76, no. 2–4, pp. 537–541, 1999.

[18] K. Efimenko, W. E. Wallace, and J. Genzer, “Surface modification of Sylgard-184 poly(dimethyl siloxane) networks by ultraviolet and ultraviolet/ozone treatment,” Journal of Colloid and Interface Science, vol. 254, no. 2, pp. 306–315, 2002.

[19] Y. Tai, M. Watanabe, K. Kaneko et al., “Preparation of gold cluster/silica nanocomposite aerogel via spontaneous wet-gel formation,” Advanced Materials, vol. 13, no. 21, pp. 1611–1614, 2001.

[20] K. V. Sarathy, G. Raina, R. T. Yadav, G. U. Kulkarni, and C. N. R. Rao, “Thiol-derivatized nanocrystalline arrays of gold, silver, and platinum,” Journal of Physical Chemistry B, vol. 101, no. 48, pp. 9876–9880, 1997.
