1. Introduction

The function of stabilizer bars in motor vehicles is to reduce the body roll during cornering. The body roll is influenced by the occurring wheel load shift and the change of camber angle. Decisive is the steering performance which may be purposely adjusted towards understeer or oversteer when designing the stabilization. The designer has to consider many factors when constructing stabilizer bars. This is of fundamental importance for successful construction and series production as well as for the quality and fatigue strength of the stabilizer bar. Stresses and deformations occurring in the stabilizer bar under load may be calculated using two different methods: analytical calculation and finite element method (FEM). If we consider the loads of stabilizer bars over the material cross-section, we immediately realize that the inner sections are not loaded, so they do not contribute to the function. Therefore, in order to reduce the weight, the tubular stabilizer bars were developed. The relationship of solid bar diameter, tube dimensions, weight saving and stress increase for a solid bar diameter of 25 mm, as shown in example Fig. 1 [1–6, 9]. Realistically, the weight saving does not exceed 45 % because otherwise the tubes become too thin-walled, and considerable manufacturing difficulties occur. The load also linearly increases with the diameter so that already for this reason natural boundaries are set. That is why a relationship of outer diameter to wall thickness of 6.5 to 7.5 should not be exceeded. In addition, provided that the stabilization is equal, approx. 6.5 % bigger outer diameters and as a consequence accordingly bigger stresses with a weight saving of approx. 40 % compared to solid bars arise. The points of force transmission in tubular stabilizer bars may be made identically to those of bars of solid material [1–6, 9]. During driving for each vehicle many factors can be determined, like acting mechanical forces, vibration, noises and in consequence determination of the degree of wear, friction and corrosion processes For proper analysis of those phenomenons has to be conducted inter alia, the strength testing and metallographic research. The material properties and metallurgical processes are very important, especially for vehicle elements which correspond
to for safety of driving. There are many new research, reports and papers about novel materials in terms of the properties and metallurgical processes.

2. Experimental materials and conditions

The stabilizer bar was hot bent at the bending table, hardened in oil and tempered (Table 1). The bending radii I, II and III (Fig. 2) were metalographically tested. The stabilizer bar was also subjected to fatigue tests. For analysis purpose the finite element method [7-9] calculation have been conducted. The stabilizer bar has broken early (it has not reached the specified figures of the load cycles). The reason of it is proeutectoid ferrite precipitations connected with the high load in the bending radii areas.

![Fig. 2. Tubular stabilizer bar – endangered areas, area of breakage](image)

The metallographic examination of fatigue fractures radial zones of stabilizer bar were carried out in the laboratory of ThyssenKrupp Bilstein Suspension GmbH (Fig. 3) using a Leitz microscope metallographic type Aristomet, coupled with Sony Videoprinter video equipment, cameras Leicaflex, foto-makroskop Wild 400 and a device for measuring surface roughness Perthometer S5P.

![Fig. 3. Metallographic laboratory of ThyssenKrupp Bilstein Suspension GmbH (source TKBiS)](image)

The fatigue tests were performed on eccentric machine for dynamic tests of stabilizer bars (Fig. 4). The number of fatigue cycles obtained during tests (Fig. 5) (until the destruction of stabilizer) was compared with the results of FEM calculation (using of dynamic fatigue calculation software DesignLife nCode HBM) (Fig. 15).

![Fig. 4. Eccentric machine for dynamic tests of stabilizer bars (source TKBiS): max. dimensions of stabilizer bar: 1100 x 350 mm, diameter of stabilizer bar: 10 – 25 mm, max. forces acting on the ends of the stabilizer bar: ± 4 kN, angle of oscillation: ± 40°, frequency: 0,5 – 4 Hz](image)

The FEM strength calculations (static calculations – Fig. 13, 14 and dynamic calculations – Fig. 15) were carried out using programs HyperWorks + Radioss (Altair) and nCode DesignLife (HBM).

On microscopic evaluation, the depth of decarburization is based on structural changes determined (according to DIN EN ISO 3887) [10]. At ferrite–pearlite structures in the decarburization edge zone a can be observed with higher ferrite content. If there is a soft annealing structure, the

![TABLE 1](image)

| Production parameter | value |
|----------------------|-------|
| tubular stabilizer bar | Ø23,5x3,6 mm |

\[\textbf{Table 1: Production parameters} \]
decarburization shall be determined by a reduction of the carbide content in the ferritic matrix.

![Graph of fatigue tests results](image)

**Fig. 5.** The results of fatigue tests carried out on the machine shown in Fig. 4

### 3. Results

Solid stabilizer bars are manufactured of heat–treated steels according to DIN 17200 and 17221. For tubular stabilizer bars primarily micro–alloyed steels such as 26MnB5 and 34MnB5 (Table 2), which may be either hot or cold bent, are used. The maximum allowable strength, thus the load is determined in the first approximation by the carbon content. In case of welded tubes the maximum carbon content is limited by the welding process. If the carbon content of an alloy is not sufficient in order to set a desired strength, the stabilizer bar manufacture must be preceded by a carbonisation process of the semi–finished tube (Fig. 6).

| TABLE 2 Chemical composition 34MnB5 [14] |
|-----------------------------------------|
| type of steel | number of steel | C   | Si  | Mn   | P |
|---------------|-----------------|-----|-----|------|---|
| 34MnB5        | 11.166          | 0.338 | 0.24 | 1.22 | 0.01 |
|               |                 | S    | Al  | N    | Nb |
|               |                 | 0.001 | 0.03 | 0.0044 | 0.001 |
|               |                 | V    | Mo  | B    | Cu |
|               |                 | 0    | 0.01 | 0.0021 | 0.01 |
|               |                 | Cr   | Ni  | Ti   |    |
|               |                 | 0.13 | 0.02 | 0.03 |   |

Proeutectoid ferrite occurs in the form of separate grains alternately with the pearlite grains (cellular structure) or at the boundaries of the pearlite grains [15]. The stabilizer bar shows the highest percentage of ferrite in the back (bending radius I). It tends to decrease towards the arms (radii II and III – Figures 8 and 9), according to the bending sequence. The percentage of ferrite, however, is generally too high. The values in the amount of 0.15 mm will not lead (Table 3) to the fatigue fracture. The values above 0.15 mm lead to the fatigue fracture and thus premature failure of the stabilizer bar.

| TABLE 3 Production parameters |
|-------------------------------|
| example / radius | decarburization | forming of steel microstructure | drawing |
|------------------|-----------------|------------------------------|--------|
| I outside of tube | the sample is over the entire cross–section: | martensite and formation - excretion of proeutectoid ferrite 0.18 mm | drawing 7 a–c |
| I inside of tube  | formation - excretion of proeutectoid ferrite 0.13 mm | martensite + proeutectoid ferrite | drawing 8 a–c |
| II outside of tube | decarburization: ca. 0.01mm | formation - excretion of proeutectoid ferrite 0.13 mm | drawing 9 a–b |
| II inside of tube  | formation - excretion of proeutectoid ferrite 0.14 mm | martensite | |
| III outside of tube | decarburization: ca. 0.01mm | formation - excretion of proeutectoid ferrite 0.08 mm | |
| III inside of tube  | formation - excretion of proeutectoid ferrite 0.08 mm | | |

Ferrite is a solid, interstitial carbon solution in iron which forms through entering of carbon atoms into octahedral voids, which are flattened, and into tetrahedral voids (Fig. 7). The fact that the carbon atom diameter is greater than the diameter of the voids causes that the solubility of carbon is low and does not exceed 0.022%. Ferrite as a separate structural constituent occurs in hypoeutectoid steels so called proeutectoid ferrite. In view of a low carbon content the properties of ferrite do not differ much from the properties of pure iron α so that $R_m =$ about 300 MPa, $80 \text{ HB}$, $A_1 = 40\%$, $K_C =$ about $180 \text{ J/cm}^2$. It is visible as a bright constituent in the metallographic specimens.

![Tempering curves of steel grade 34MnB5](image)

*Fig. 6. Tempering curves of the steel grade 34MnB5 [5]*
Fig. 7. The microstructure formation – inside and outside of the tube (cross section I) [11, 12, 13]:

a) radius I / inside bending radius, tube outside, etched with 1% HNO₃, 200:1,

b) radius I / inside bending radius, tube outside, etched with 1% HNO₃, 50:1,

c) radius I / core, etched with 1% HNO₃, 200:1,

d) radius I / inside bending radius, tube inside, etched with 1% HNO₃, 200:1,

e) radius I / inside bending radius, tube inside, etched with 1% HNO₃, 500:1,

f) precipitates of proeutectoid ferrite

Fig. 8. The microstructure formation – inside and outside of the tube (cross section II) [11, 12, 13]:

a) radius II / inside bending radius, tube outside, etched with 1% HNO₃, 500:1,

b) radius II / core, etched with 1% HNO₃, 500:1,

c) radius II / inside bending radius, tube inside, etched with 1% HNO₃, 500:1

Fig. 9. The microstructure formation – inside and outside of the tube (cross section III) [11, 12, 13]:

a) radius III / inside bending radius, tube outside, etched with 1% HNO₃, 500:1,

b) radius III / inside bending radius, tube inside, etched with 1% HNO₃, 500:1

Figures 2 and 10 show a hot bent, highly stressed stabilizer bar. The applied 34MnB5 steel is an alloy steel with low carbon content. Despite the very short time of the bending process, the formation of proeutectoid ferrite occurs. Fig. 11 shows an isothermal transformation diagram for the 30MnB5 steel which has properties comparable to the 34MnB5 steel. The mechanism of ferrite formation in alloy steels with low carbon content (up to 0.45%) is shown in Fig. 12. The desired martensite transformation with bypassing the pearlite and bainite transformation areas does not occur here. The steel reaches a lower hardness (yellow line) about 318 HV ($R_m = 1022\, \text{MPa}$)

Fig. 10. Area of breakage – warm bending of tubular stabilizer bar in the bending table [12, 13]

Fig. 11. TTT diagram (time temperature transformation) for steel 30MnB5 [14]

Fig. 12. TTT diagram (time temperature transformation) for alloy steels with carbon content below 0.45% [15]
4. Discussion

The breaking point is not located where the greatest equivalent stresses occur (Fig 13 c, d). The stabilizer bar breaks there where the greatest proeutectoid ferrite precipitations occur (Fig. 2, bending radius I and Fig. 13b). The maximum equivalent stress at the breaking point is about 714,05 MPa. So the combination: high stresses connected with the ferrite precipitations resulted in the early destruction of the stabilizer bars. For tube stabilizer bars, hot bent with mechanically high stress, steel 34MnB5 is not recommended. The reason is ferrite formed in the radial zones and will lead to a fatigue fracture. In the case of stabilizer bars with mechanically low stress, formation of ferrite in the radial zones will be omitted.

It is reasonable to conduct more investigation for such important element in car vehicle. Taking into consideration all processes, starting from manufacturing [16-20], regular exploitation [21-24] and repairs [25-32].

Nowadays many of monitoring system allow observing of functionality and parameters of different elements [33-40].

5. Conclusion

In the process of tube bending the outer curve of the tube is elongated and the inner curve is upset forged. Furthermore, circumferential stress occurs. The consequences of this complex triaxial stress and deformation condition are displacements of the non–stretched layer towards the bending center, in circumferential direction – elongations of the stretched side and upsetting of the compression side, changes in the cross–sectional shape (ovalization) and an irregular distribution of cold work hardening over the cross–section and unrolling the bend in relation to longitudinal axis.

It is important that the intended weight reduction is not greater than 40 to 45 % of the weight of a comparable solid stabilizer bar. Weight savings exceeding beyond that lead to no longer acceptable wall thicknesses, thus to further loads which are caused by the warps of the tube. Such arch stresses have an extremely negative impact on the lifespan of the element. The outer and inner diameters can be calculated by the following equations [1–6]:

\[ d_{\text{tube}} = d_{\text{solid}} \sqrt{\frac{1-k^2}{2k}} \]  
\[ d_{\text{a,tube}} = d_{\text{solid}} \sqrt{\frac{1+k^2}{2k}} \]

where:
\( d_{\text{tube}} \) – inner or outer diameter of tube
\( d_{\text{solid}} \) – diameter of solid stabilizer bar
\( k \) – weight fraction of the tube e.g. \( k = 0,6 \)

Fig. 13. Finite Element Method (FEM) computations – central bending radii, bearing area (Fig. 2):
a) equivalent stresses according to Tresca (913,6 MPa),
b) equivalent stresses according to v. Mieses (714,05 MPa) – breakage / bending radius,
c) and d) equivalent stresses according to v. Mieses (891,8 MPa) – bending radius II,
d) equivalent stresses according to v. Mieses (835,1 MPa) – bearing area

A proper construction and the selection of parameters influence the strength properties, the weight, durability and reliability as well as the selection of an appropriate production method. The selection of the manufacturing process has a fundamental impact on the quality and durability of the stabilizer bars.
Fig. 14. Equivalent stresses: a) analytical calculations, b) tension curve created with HyperWorks (FEM)

Fig. 15. Durability analysis – created with HBM nCode DesignLife

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