Dielectric Mie voids: confining light in air

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Abstract
Manipulating light on the nanoscale has become a central challenge in metadevices, resonant surfaces, nanoscale optical sensors, and many more, and it is largely based on resonant light confinement in dispersive and lossy metals and dielectrics. Here, we experimentally implement a novel strategy for dielectric nanophotonics: Resonant subwavelength localized confinement of light in air. We demonstrate that voids created in high-index dielectric host materials support localized resonant modes with exceptional optical properties. Due to the confinement in air, the modes do not suffer from the loss and dispersion of the dielectric host medium. We experimentally realize these resonant Mie voids by focused ion beam milling into bulk silicon wafers and experimentally demonstrate resonant light confinement down to the UV spectral range at 265 nm (4.68 eV). Furthermore, we utilize the bright, intense, and naturalistic colours for nanoscale colour printing. Mie voids will thus push the operation of functional high-index metasurfaces into the blue and UV spectral range. The combination of resonant dielectric Mie voids with dielectric nanoparticles will more than double the parameter space for the future design of metasurfaces and other micro- and nanoscale optical elements. In particular, this extension will enable novel antenna and structure designs which benefit from the full access to the modal field inside the void as well as the nearly free choice of the high-index material for novel sensing and active manipulation strategies.

Introduction
Resonant optical phenomena in metals and dielectrics have found profound applications in many fields1. The nanoscale confinement allows for unprecedented control of light-matter interaction at surfaces and interfaces2, both in manipulating and controlling the flow of light3–6 but also in enhancing the interaction with localized nanoscale systems such as emitters or sensor elements7,8. Resonant phenomena are usually associated with radiative and intrinsic loss channels, which are detrimental in many systems. In particular, metals show strong intrinsic losses. Thus, dielectric systems recently came into the focus of attention as they promised lower loss, higher degrees of flexibility with respect to tuning the interplay of different resonances9–12, as well as fabrication strategies closer to industrial standards. The optical response of these systems is best understood in terms of the well-known Mie theory13, mostly related to the study of optical properties of high-index dielectric nanoparticles in nanophotonic systems14–18. They form universal building blocks for optical metasurfaces19–24 which are used in the manipulation25–31, routing32, and confinement of radiation33, and thus trigger the start of the “Mie-tronic” era34. However, even in these systems, intrinsic loss plays an important role: The confinement takes place inside the high-index materials, most modal intensity is thus located inside the material. While this is of less concern in the near- and mid-infrared wavelength ranges, it becomes crucial for wavelengths in the visible or even the UV spectral range, where the intrinsic material loss hinders or even prevents the observation of resonant modes.
In this article we experimentally implement an elegant and powerful alternative route utilizing high-index materials, namely, the resonant confinement of light in air. In general, confinement in high-index dielectrics occurs due to a finite reflectance at the interface of the high-index material and air. For solid particles, the mode is thus localized within the high-index material. We theoretically and experimentally demonstrate that also in the inverted case of an air void inside a homogenous high-index medium, localized optical modes emerge, which are confined within the nanoscale low-index void by virtue of the finite reflectance at the material discontinuity. While the basic concept has been found earlier\cite{35,36}, these structures have not been implemented for nanophotonic applications, despite their superior properties. We show that these void modes are predicted by Mie’s theory and bear close resemblance with the Mie modes of a high-index sphere, yet showing subtle, but significant differences. The novelty of our work, compared to the state of the art, is thus the experimental implementation of a new building block which adds true novel functionality to the field: We experimentally demonstrate the resonant confinement of UV radiation, observing up to seven higher-order resonant modes in the UV. This astonishing feature is enabled by the confinement of light in air, rendering the entire modal volume accessible to be used and manipulated. To date, there is no other resonant nanophotonic building block which can perform this task.

We experimentally realize Mie voids by focused ion beam milling of cylindrical holes into bulk silicon wafers. Examples are shown in Fig. 1. Panel a depicts a scanning electron microscopy (SEM) image of a random arrangement of voids with varying diameter and depth. The optical microscope image (imaging setup: panel f) in panel b shows the distinctively different resonant optical scattering response of the voids, clearly depending both on diameter and depth. This underpins an additional strength of our FIB implementation: In contrast to the conventional fabrication method via reactive etching of silicon disks, we utilize the intrinsic depth variation capability as an additional degree of freedom. This feature is best visible for the chain of voids (panel c: optical microscope image, panel d: top view SEM image). The focused ion beam cut (shown in panel e) demonstrates the significant differences in size and depth, manifesting themselves in a distinct colour impression of each void (see microscope image in panel c). The dependence on diameter and depth clearly rules out trivial interference effects. The distinct colour of each void in the random as well as chain arrangement rules out grating phenomena, as grating colours would not vary across the grating and would only be determined by the grating constant (here 900 nm).

Results

As a first step, we will solve Mie’s theory for the case of voids in a high-index surrounding, as a confirmation of our experimental findings. In general, it is not possible to analytically solve Maxwell’s equations for propagation and interaction of light for any given situation. Mie, however, was able to show that the scattering problem for a spherical inclusion of radius $R$ composed of a material with refractive index $n_1$ in a surrounding medium of refractive index $n_2$ can be solved analytically by a decomposition of the incident and outgoing waves into vector spherical harmonics. The strength of Mie’s theory lies within these analytical solutions, which allow for intuitive insight into the behaviour of the system, e.g., the dependence on...
geometrical parameters or the complex dielectric function of the materials (which includes intrinsic loss). They also allow to deduce the near- and far-field behaviour. Interestingly, in the majority of cases, researchers have studied the case of high-index spheres in a low-index surrounding, in particular for the implementation of high-index dielectric nanophotonics. Mie’s solutions, however, describe a broader scenario as the model places no restrictions on the relative or absolute values of the dielectric function inside and outside the sphere, which opens a second, symmetric parameter space, namely low-index voids with refractive index \( n_v \) placed inside a high-index dielectric host medium \( n_h \) (\( n_v > n_h \)). While a similar phenomenon is known for metallic structures by virtue of Babinet’s principle\(^\text{[47-49]}\), in dielectric systems the full strength of this symmetry has not been explored or experimentally implemented. Optical resonances in textured, semi-periodic dielectric void systems of corrugated silicon surfaces have been studied and resonances associated with voids have been found, which were utilized for absorption enhancement\(^\text{[40]}\) as well as light manipulation\(^\text{[41]}\).

In nanoporous, periodic gold surfaces void-based resonances were found, which are interpreted as combinations of Mie void resonances and plasmonic phenomena due to the presence of the metal\(^\text{[42,43]}\). Also, air bubbles in water have been studied with ray-optical as well as Mie scattering methods. However, as the refractive index contrast between the air bubbles and water is only on the order of 0.35, resulting in quality factors on the order of 1, no strong modal confinement can be observed (please also see Fig. S3 in the Supporting Information). Additionally, bubble sizes are often in the several to hundreds of micron region (placing them into the realm of ray optics)\(^\text{[44,45]}\). Moreover, voids in host media have been investigated in the realm of effective material properties\(^\text{[46]}\). However, the confinement to the low-index void material and the fact that the surrounding host dielectric high-index material and its generally significant loss play a minute role for the properties of the modes has not been exploited and in particular not been experimentally implemented. It should be noted, that this behaviour is in stark contrast to inverse Babinet-type structures known in metallic systems, which are subjected to significant loss in the surrounding metal due to currents and fields inside the metal. Arguably, the concept is related to hollow-core photonic crystal fibres\(^\text{[47,48]}\), which also confine light to the air-filled core region. However, the physical concept is different for two distinct reasons: The modes in the fibres are propagating ones and are not localized into nanoscale volumes. Additionally, the confinement to the air core is enabled by dispersion engineering inside the photonic crystal fibre, rather than Fresnel reflection.

First, we notice that the denominator of the scattering amplitudes in Mie theory determines the eigenfrequencies of the resonant states, that is, the TM and TE modes in a spherical inclusion, obeying the following relations

\[
\begin{align*}
\text{TM modes} & \quad \frac{\psi_j(n_e k_0 R)}{\psi_j(n_v k_0 R)} = \frac{n_v \xi_j(n_v k_0 R)}{n_e \xi_j(n_e k_0 R)} \\
\text{TE modes} & \quad \frac{\psi_j(n_e k_0 R)}{\psi_j(n_v k_0 R)} = \frac{n_v \xi_j(n_v k_0 R)}{n_e \xi_j(n_e k_0 R)}
\end{align*}
\]

The prime denotes derivatives with respect to the argument, while \( k_0 = \omega / c \), and \( \psi_j \) and \( \xi_j \) are the Riccati-Bessel functions of order \( l \), with \( \psi_j(x) = x j_l(x) \) and \( \xi_j(x) = x h_l(x) \), where \( j_l \) and \( h_l \) are the spherical Bessel and outgoing spherical Hankel functions, respectively. We assume non-magnetic materials \( (\mu = 1, \varepsilon = 1, n = \sqrt{\varepsilon}) \). Solving these relations, one can calculate the resonant wavelength, linewidth, and dispersion of the modes with material dispersion of refractive indices \( n_v \) and \( n_e \), and associated material losses, as well as the near- and far-field behaviour.

In Fig. 2 we display the analytical results for the eigenmodes and dispersion relations assuming silicon and air as the host and surrounding media. Silicon is a prime candidate for high-index dielectric nanophotonics due to its abundance, low cost, and well-established industry-level fabrication strategies\(^\text{[49,50]}\). Figure 2a depicts the analytical results for a silicon sphere in air. We calculate the four lowest energy eigenmodes and show one spatial cross section for each (all spatial slices of the modal field plots can be found in Fig. S1 in the Supporting Information). We were able to confirm the existence of an electric and magnetic dipolar modes as well as the electric and magnetic quadrupolar modes\(^\text{[42]}\). Figure 2b depicts the case of the dielectric Mie void. In spite of the very different geometry of sphere and void, the symmetry properties of both structures are rather similar. The analytical solutions of Mie sphere and Mie void show a significant resemblance of the near fields for all four fundamental modes. In spite of these similarities, there are subtle, yet important differences in the field distributions: In case of the silicon spheres, the modes are pinned to the silicon-air interface and most of the field is confined inside the high-index material, yet, also significant field components are extending into the lower index surrounding. In contrast, the Mie void modes show stronger confinement within the air void and barely any modal components extending into the silicon host. This is an intriguing phenomenon for two distinct reasons: The mode barely suffers from the intrinsic silicon loss while it simultaneously allows access to the entire modal field inside the air void. Contrary to the Mie sphere, one can thus push the resonant energies far into the visible and UV spectral range where silicon typically exhibits large absorption. We note that here we normalize the mode fields for Mie spheres and voids.
identically using the exact analytic normalization procedure for open resonators\cite{51}.

Next, we compare the dispersion relations for the modes of Mie sphere and Mie void, to further underpin our interpretation. The graphs in Figs. 2a, b depict the radius dependence of the resonant wavelength of all four modes within the application-relevant wavelength regime from 300 nm to 850 nm. In case of the Mie sphere (Fig. 2a) the dispersion relations are strongly influenced by the intrinsic silicon dielectric material dispersion, evidenced by their bending and the absorptive features around 380 nm. The Mie void modes exhibit perfectly linear dispersion (Fig. 2b), indicating that the modes are indeed fully localized in air. Moreover, the Mie voids need to be significantly larger compared to the Mie sphere in order for the resonant wavelength of the modes to be in the same spectral range. The reason lies with the significant refractive index difference between silicon and air, which are on the order of

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Fig. 2 Mie’s theory for silicon particles in air and air voids in Si. a Field profiles and dispersion relations for localized Mie modes of a solid silicon Mie sphere in air and b, of a Mie void, which is, a spherical void inside an isotropic silicon environment. In both cases, Mie theory predicts the existence of localized modes (electric and magnetic dipole as well as quadrupole and higher-order modes). In spite of the inversed geometry, the modes show strong resemblance, yet, differ in the degree of confinement to the silicon and air, respectively. The dispersion relation of the modes of the silicon sphere is strongly influenced by the spectral dependence of the silicon refractive index and absorption (see c). For the Mie voids, in contrast, we observe perfectly linear dispersion, showing that the modes are fully localized in air without any noticeable influence of the silicon dispersion. c Real part $n$ and imaginary part $\kappa$ of the refractive index of silicon (upper panel) in comparison to the properties of the sphere and void modes (middle and lower panel, respectively)\cite{64}. The graphs show the quality factors of each of the four fundamental modes for the sphere and void vs. the resonant wavelength of each of these modes. The dependence of the quality factor on the resonant wavelength is calculated by sweeping the sphere and void size. In case of the sphere (middle panel), one observes a drastic reduction of the quality factor for resonant wavelength approaching 400 nm and no modes are observed for wavelengths < 380 nm. This behaviour can be explained by the material losses in silicon, depicted with a dashed line showing the material quality factor $n/2\kappa$. In case of the void (lower panel) the modes are confined within the air void. Consequently, the losses in silicon do not lead to a damping of the modes but rather result in a stronger confinement of the mode in the void by virtue of increased Fresnel reflection coefficient at the air-silicon interface (increased refractive index and increased absorption). Consequently, one can excite strongly confined modes in the deep-blue and UV spectral ranges with quality factors up to 35.
n_{\text{silicon}} \approx 4$ and $n_{\text{air}} \approx 1$. Moreover, we also observe a rearrangement of the energetic ordering of the modes. The fundamental eigenmode of the Mie sphere is the magnetic dipolar mode, whereas in case of the Mie void it is the electric dipolar mode.

The aforementioned results have emphasized the key role of the material dispersion on the optical and resonant properties. To further elucidate this behaviour, we plot in Fig. 2c the real and imaginary part of the refractive index of silicon (upper panel) in comparison to the properties of the Mie sphere and void modes (middle and lower panel, respectively). The quality factors of each of the four fundamental modes are calculated and plotted against the resonant wavelength of each of these modes—the middle and lower panels thus depict the quality factors of the modes for a radius sweep of sphere and void, respectively. Supporting Information Fig. S2 shows the data directly plotted as a function of size and confirms that the mode quality factor increases with the resonant frequency and mode order for Mie spheres and voids of fixed radius. More information on quality factor evaluation for dispersive media can be found in Materials and methods. In case of the sphere, one observes a drastic reduction of the quality factor for resonant wavelengths approaching 400 nm. More striking, no modes are observed for wavelengths shorter than ~380 nm. This behaviour can be explained by the optical loss in silicon. The dashed line indicates the material quality factor $n/(2\kappa)$, which is a measure for the relative contributions of dielectric response and material loss. As one can see, all modes in silicon are restricted by this quantity—no modes can be excited at frequencies beyond this line as intrinsic damping dominates over the resonant behaviour. In case of the voids, plotted in the lower panel, we observe a completely different situation. For longer wavelengths, the quality factors are on the order of 10, comparable to the ones observed for the dipolar Mie sphere modes. At shorter resonant wavelengths, one finds an increase in the quality factors, in particular below the cut-off observed for the Mie sphere. As the modes are confined in the air void, the loss in silicon does not lead to a damping of the modes. On the contrary, the increase in the refractive index (real part $n$) at ~380 nm and the increase in the loss/absorption (imaginary part $\kappa$) lead to a stronger confinement of the mode to the void by virtue of increased Fresnel reflection at the air-silicon interface (note that the peaks in the quality factor dispersion coincide with the material's resonant features in $n$ and $\kappa$ for silicon, displayed in the upper panel in Fig. 2c). Consequently, the inverted geometry of the Mie void and the confinement to air instead of the high-index lossy material reveals its full strength: The structure supports strongly confined modes even in the UV spectral region with quality factors up to 35 without suffering from material loss. Similar behaviour has been observed in hollow-core photonic crystal fibres that can, theoretically, exhibit extremely low attenuation due to the guiding of light inside the air-filled core.

In the Supporting Information Figs. S3 and S4 we explore this behaviour further for varying material parameters (“artificial materials”). We observe a stronger confinement and thus larger quality factor of the Mie void modes for increasing refractive index and increasing loss as well as a full damping of the Mie sphere modes for increasing loss, underpinning the above interpretations. We also observe very weak dependence of Mie void mode resonant wavelengths on the complex dielectric function of the host material, which makes them robust to inevitable material parameter fluctuations due to fabrication tolerances.

We note that the conventional method of modal analysis by using scattering or extinction cross-section to evaluate mode wavelength, linewidth and quality factor are not applicable for the case of single Mie voids embedded in homogeneous and isotropic high-index media. The non-resonant background response due to high-index surroundings is strong and smears out the mode’s resonant features in the spectrum of the Mie voids. To eliminate the effect of non-resonant oscillations for single voids in a substrate, we consider evaluating the reflectance for a silicon substrate with individual voids located in it. By proper normalization of the reflected signal to the bare response of the system without voids, the effect of background oscillations can be eliminated. Therefore, the reflectance spectra can be used for extraction of the mode’s resonant properties, as shown below for both experimental and calculated data.

Our ansatz for an experimental realization is shown in more detail in Fig. 3. Voids inside bulk silicon cannot be accessed optically in the visible wavelength range due to absorption in the host material. For an experimental implementation one needs to bring them close to the surface, as already shown in Fig. 1. We utilize FIB milling into bulk silicon wafers to realize Mie voids as conically shaped voids at the silicon-air interface (see Materials and methods section for details). Figure 3a depicts SEM images of our Mie voids, taken for the structures that are spectrally characterized. Reflectance spectra are measured from arrays of $10 \times 10$ elements at a period of 900 nm in both directions (Fig. S5 in the Supporting Information demonstrates experimentally that the periodicity and grating arrangement have negligible influence on the optical properties). The SEM images show voids with a diameter of ~610 nm and a depth of ~410 nm, illustrating the slightly inclined side walls, which are inherent to the milling process. The lowest image depicts a FIB cut of holes with 780 nm diameter at a tilting angle of 80° and underpins the conical shape.

Figure 3b shows the evolution of the experimental and simulated reflectance spectra of periodic void arrays with
increasing diameter from ~450 nm (uppermost spectrum) to ~810 nm (lowest spectrum) with a depth of ~450 nm (the detailed structural parameters, which are also used in the simulation, can be found in Fig. S6 in the Supporting Information). Arrangement in ordered arrays is done for the sake of simplicity of spectroscopic measurements while the resonant features originate from a single void. For the smallest void, we observe one resonant feature in the reflectance spectrum with strong localization in the void. This modal signature undergoes a spectral red-shift for increasing diameter. Additionally, a second feature appears at shorter wavelength, also undergoing a red-shift with increasing size. We note that these features come about due to the combination of the resonant scattering of the Mie voids and the large off-resonant reflectance of the bare silicon substrate on the order of 40%. The experimental spectra show excellent agreement with full-wave simulations of periodic arrays (see Materials and methods for details). We also performed reflectance calculations for single voids in a silicon substrate, which are shown in Fig. S7. One can see pronounced resonant features that confirm that reflectance analysis for single voids can be used to extract resonant properties from the spectrum along with the eigenmode simulations. The excellent agreement between experiment and simulation again points to the localized nature of the observed modes. In order to elucidate the nature of the observed modes, we show electric field profiles for the second largest structure (see panel c) at wavelength points I and II at which we observe the modal signatures clearly in the field plots. The three cross-sections for both modes at the selected wavelengths are displayed in Fig. 3c. Comparing the field distributions to the analytical Mie void modes depicted in Fig. 2b and Fig. S1, we observe significant resemblance. Despite the fact that the experimental Mie void is bound to the interface and has a truncated cylindrical shape rather than a spherical one, the simulated mode I shows excellent agreement with the fundamental electric-dipole Mie mode of the spherical void. For the higher-order mode II, we observe reasonable resemblance with the next higher-order magnetic-dipole mode of a spherical void. A detailed comparison of the modes, including the modal distributions obtained from eigenmode simulations, can be found in Fig. S8. Here we show that the modes observed in the truncated conical voids can indeed be understood in terms of the leading order modes of a spherical Mie void. The calculation results show two additional benefits of Mie void modes over conventional Mie resonances of dielectric particles: The mode wavelength and field profile are robust to geometry change (from spherical to cylindrical and truncated conical shapes). Moreover, the void mode characteristics change weakly by going from an idealized void in a homogeneous environment to a realistic void in a substrate, which
differs strikingly from the properties of Mie resonances of silicon nanoparticles requiring a low-index substrate.

The results shown in Fig. 2 predict resonant behaviour down to the UV spectral range. In order to study the optical response of the Mie voids over a large range of sizes and depths and thus cover the spectral range from the UV to the near infrared we utilize an elegant imaging technique which allows to capture a large set of information at once. A fine variation of the geometrical parameters will enable the simultaneous observation of several mode orders when imaging the optical response at individual wavelengths. Depending on the diameter and the depth, the voids will support modes of different order, which all have the same resonant wavelength. The white light microscope image of such a sample for “geometrical spectroscopy” is displayed in Fig. 4a, while panel b shows selected SEM images. We vary the diameter and depth of the voids in 36 steps each, with diameters between ~330 nm and 750 nm and depths between ~20 nm and 1100 nm (all parameters can be found in the Supporting Information). The images are taken for the smallest diameter voids for smallest and highest dose (shallowest (iii) and deepest (i) voids, respectively) as well as for the largest void diameter for smallest and highest dose (shallowest (iv) and deepest (ii) voids, respectively). c Experimental and simulated monochromatic images of the void array for the indicated wavelengths (displayed as vertical labels on the left). In the experiments, band pass filters at the specified centre wavelength are used to limit the wavelength range. With decreasing wavelength, an increasing number of modes can be observed. Due to the shorter wavelength, more and more higher-order modes can be resonantly excited. It is noteworthy, that this behaviour depends both on the depth as well as the diameter, ruling out trivial interference effects. In the UV spectral range at 265 nm or 4.7 eV, one experimentally observes seven distinct higher-order modes. The simulated optical response is in good agreement with the experiment.

The white light microscope image in panel a exhibits smooth and steady transitions in the colour impression of the structures, underpinning the fine parameter variation. One can clearly observe resonant modes which scatter light at distinct wavelengths. The scattering behaviour depends both on the depth and diameter, ruling out a trivial interference phenomenon. Additionally, one can identify regions of identical or similar colour impression, showing that modes of different order are observed, which each scatter at a similar wavelength. This behaviour is particularly well visible for the blue and red stripes.

The white light impression already points to the measurement strategy: In order to identify modes of different order at the same wavelength, we image at distinct wavelengths, separating the optical response of each “colour”. We utilize band pass filters between 265 nm and 800 nm and take monochromatic images of the array, as shown in Fig. 4c. These images thus directly visualize the resonant scattering behaviour of the structures at the respective wavelength. As mentioned, due to the variation of diameter and depth we directly visualize the dispersion of the modes.
with respect to these two parameters. Each experimental image is accompanied by a simulated one. The simulated optical response is based on the experimentally determined diameters and depth of the voids, without any further assumptions (see Materials and methods for details). In each image we observe strong modulation in the recorded far-field behaviour, which are caused by resonant scattering of the Mie voids, modifying the off-resonant reflectance of the bare silicon. Each modulation is thus associated to a resonant mode. With decreasing wavelength an increasing number of modulations and thus modes can be observed: For the image at 800 nm we only observe one mode, for 550 nm three modes can be identified, and for the shortest wavelength at 265 nm or 4.7 eV we identify seven modes. Due to the shorter wavelength an increasing number of higher-order modes can be resonantly excited. Overall, the simulations are in good agreement with the experiment.

As the single Mie voids displayed in Fig. 1b show brilliant and naturalistic colours in their scattering response, our Mie void concepts holds great promise for nanoscale colour generation and colour printing\textsuperscript{53,54}. We explore this application in Fig. 5. Nanoscale colour printing that utilizes resonant scatterers depends on several key features, which are the quality factor, the scattering amplitude, and the tunability\textsuperscript{55–57}. The resonances modify the reflected incident white light and are thus responsible for the colour impression. If the resonances are spectrally too narrow, the colours appear artificial and neon-coloured; if they are too broad or the scattering amplitude is too small, the colours are dull. As the image in Fig. 1b as well as the spectra in Fig. 3c clearly show, our Mie voids have ideal properties for brilliant and vivid colour generation in reflection. Figure 5a depicts a selection of our “colour catalogue” which has been used to generate our Mie void colour prints. For illustration, we show the colour impression of arrays of 10 × 10 elements with 900 nm period as well as SEM images for two cases: In the upper row voids of constant diameter and increasing depth, in the lower row for constant depth and increasing diameter. Both parameters allow to tune the colour impression over the entire visible wavelength range (please also refer to the large sweep over diameter and depth shown in Fig. S9 in the Supporting Information). We find that similar colours can be obtained from different combinations of diameter and depth, while the volume of the void determines the absolute scattering efficiency and thus the saturation of the colour. Supporting Information Figs. S10–S13 as well as Table S7.3 show the optical appearance, reflectance spectra, and geometrical parameters of the full colour catalogue used. Details to the colour print generation can be found in the Materials and methods section.

As an example we choose the painting “Improvisation No. 9” painted in 1910 by Russian artist Wassily Kandinsky from the collection of the Staatsgalerie Stuttgart, Germany. Figure 5b depicts a section of the original painting, its experimentally colour-printed version, as well as normal and tilted view SEM images of the constituent Mie voids. The period is kept constant at 900 nm, ruling out any grating effect. Each pixel is thus composed of one Mie void each. As a consequence, even in the SEM image the contours and different components of the painting can be clearly identified. Additionally, it is obvious that the diameter and depth of the Mie voids serve as colour tuning knobs. Comparing the original painting and its colour-printed counterpart, we find that the colour reproduction of the Mie-void colour printing is excellent, which is aided by the availability of “white” and “black” hues, that enable us to achieve large dynamic range and good contrast.

Figure 5c finally depicts the colour print of the entire painting “Improvisation No. 9” next to two human hairs for size comparison. The image has a size of 200 pixels by 200 pixels, a side length of 180 μm, and thus a resolution of 36.000 dpi. The colours are vivid and naturalistic and stand out against the unstructured bare silicon background of the wafer. As the colours are generated by voids in a silicon surface, the structure is remarkably long time stable due to the inertness and hardness of silicon.

**Discussion**

In summary, we have introduced the concept of dielectric Mie voids. We have shown that Mie’s theory predicts localized optical modes for low-index voids in a high-index surrounding, with the main benefit in a negligible influence of the surrounding high-index material and its high loss on the properties of the modes. In striking difference to conventional all-dielectric nanoparticles, Mie-void mode properties are very weakly dependent on the void geometry and substrate parameters. Moreover, they do not require any periodic arrangement for creating high-Q resonances. This allows to push the field of high-index nanophotonics into the blue and UV spectral range, significantly expanding the design parameter space for dielectric nano- and micro-optical elements for the future design of metadevices.

Further experimental work in this direction can make use of other high-index materials, the combination of Mie spheres and Mie voids, as well as the combination of metallic plasmonic systems with dielectric ones, in the visible and UV regions. Mie voids are also particularly well suited for optical sensing\textsuperscript{58} as well as trapping experiments and can utilize chiral structures\textsuperscript{59,60}. Mie voids could also be used in hybrid systems where quantum emitters are coupled to the Mie voids which act as local nanoantennas. This should work particularly well in the blue and UV range, where many electronic resonances are located. Filling of a quantum emitter into the voids as well as using emitters in the void’s host medium can be envisioned, for example excitons in gallium nitride or gallium arsenide, or defect centres in diamond or silicon carbide. Mie voids also hold
great promise for reprogrammable structures, switching, and active manipulation due to the possibly to fill the mode volume in the void with polymers or dielectrics. Moreover, Mie void modes are also expected to contribute to the optical, electronic, or acoustic properties in porous media, foams, glasses, dielectric and metallic networks.

Materials and methods
Structure fabrication and characterization
We utilize the Raith ionLine Plus (now Raith Velion) as a high-resolution direct structuring tool. Our Mie voids do not only require high-resolution structuring but also place significant demand on the beam and tool stability to ensure accurate and reproducible results. By virtue of a multi-species column, we are able to use singly as well as doubly charged silicon and gold ions at a fixed acceleration voltage of 35 kV for focused ion beam milling (FIB). FIB structuring is an intrinsic grey-scale lithographic process and thus offers us to vary size and depth on the same substrate. Due to larger beam current and sputtering rate we use singly charged gold ions at a beam current of ~150 pA and a 30 µm beam-forming aperture. The step size is 20 nm, circles are exposed in circular outward mode. The base dose value is 1 nC/µm². All dose values given in the manuscript are to be understood as multipliers of this basic dose value. We use standard 4'' silicon wafers, which we cleave into smaller pieces. Before structuring, they are cleaned with acetone and isopropanol in an ultrasonic bath. Scanning electron microscope images are taken with a Hitachi S-4800. Optical images of the arrays as well as the art work are taken with a Nikon Eclipse LV100ND with a 20x NA 0.45 objective at fully closed A stop. White balance is taken from the bare silicon wafer.

Spectroscopy
Reflectance spectra of ensembles of Mie voids were recorded using a custom-built microspectroscopy setup based on an inverse microscope (Nikon TE2000-U) and a Princeton Instruments grating spectrometer (SP2500i) with a Peltier-cooled CCD camera (PIXIS 256E). The sample is placed on a motorized XY stage (Märzhäuser) and illuminated by a fibre-coupled laser-driven white light source (Energetiq EQ-99XFC). A high-magnification objective (Nikon TU Plan Fluor ELWD 100x, NA 0.9) is used to observe the relatively small void arrays. The light source is projected onto the sample via a custom-built Köhler illumination path, which is designed to only illuminate a small range of incident angles (near-normal incidence). The microscope is connected to the entrance slit of the spectrometer via a modified 4-f setup, which contains an aperture in the microscope image plane (field stop) as well as in the intermediate Fourier plane (aperture stop). Because the sample reflectance can be very close to zero, both of these apertures are closed to block any light scattered from

Fig. 5 Colour printing using Mie voids in a silicon substrate at 36.000 dpi. Due to the ideal quality factor and thus linewidth of the Mie void modes, the structures show brilliant and naturalistic colours in reflection. Aided by the large interaction cross section, one Mie void is sufficient for one pixel. This gives our printing method a pointillistic touch, being completely different from grating-based approaches. a Selected sizes and depth of Mie voids and their colour impression in an optical microscope. b Detail taken from the painting “Improvisation No. 9” by Wassily Kandinsky (Staatgartlerie Stuttgart). The top left depicts the original artwork while the lower left shows an optical microscope image of the colour-printed image. In the SEM image on the right, one can clearly identify the image as the pixel size is unchanged. In order to gain access to the full colour space, the diameter as well as the depth of the Mie voids has been varied, which is particularly well visible in the tilted SEM image. c Fully reproduced “Improvisation No. 9” in a silicon substrate. The image consists of 200 pixels x 200 pixels and has a side length of 180 µm, corresponding to 36.000 dpi resolution. The image is shown next to a human hair for size comparison.
the substrate around the void ensemble as well as light scattered/diffracted into larger angles. In this way, only the zeroth-order reflectance is measured, which can be directly compared to numerical calculations. The reflectance is extracted by normalizing the measured reflected signal of a void ensemble to that of the bare silicon substrate and multiplying by the known reflectance spectrum of silicon.

UV and VIS imaging
The fibre-coupled light source DH-200 from Mikropack combines a tungsten and a deuterium lamp and provides intensity between about 250 nm to 1100 nm. The output of the fibre is collimated using a $f=35$ mm UV fused silica plano convex lens (Thorlabs). A 50:50 beamsplitter (BSW20, Thorlabs) is used to direct the light to the sample and is focused with an $f=35$ mm UV fused silica plano convex lens which also recollects the reflected light. An aperture before this lens is used to reduce the NA. An $f=50$ mm lens is used to focus the light onto an imaging spectrometer (Acton 2500) with attached Peltier-cooled CCD camera (Pixis 256E). The wavelength range of interest is set with band pass filters (Thorlabs (T), Edmund Optics (EO)) with FWHM of 10 nm (for 265 nm (EO), 313 nm (EO), 350 nm (T), 450 nm (T), 500 nm (T), 550 nm (T), 600 nm (T), 700 nm (T), 750 nm (T), 800 nm(T)) and FWHM of 40 nm (for 400 nm (T), 650 nm (T)).

Generation of art work
In the first step, the image is resized to the desired number of pixels. Afterwards, we read the colour values of each pixel of the original image in the CIELAB colour space. Subsequently, the individual LAB values for each pixel are compared to the colour appearance in the CIELAB colour space of the elements in our colour catalogue, shown in Fig. S4 in the Supporting Information. The colour difference to all colours in the colour catalogue is calculated via the CIE94 definition. For each pixel of the original image we choose the colour catalogue array with the smallest respective difference. As each array of the colour catalogue is associated with a specific void diameter and depth and thus with specific structuring design parameters, a GDSII structuring file can be created.

Simulations
Eigenmodes and Scattering Problem. For numerical simulations of the eigenmode spectra of single voids (Fig. 2 and Supporting Information Figs. S1 to S4 and S8) we use (i) a custom Python code based on deriving the analytical eigenfrequency from Eqs. 1, 2 using the argument principle method, and (ii) the finite-element method eigenmode solver in COMSOL Multiphysics. For simulations of their scattering spectra (Fig. S7) we use the frequency-domain solver in COMSOL Multiphysics. The values for the refractive index for silicon in Fig. 2 are shown in the upper panel of Fig. 2c and are taken from ref. 64. The refractive index and extinction in supporting Figures is specified for each case in the caption. For the scattering simulations, the excitation is a plane wave propagating along the Z-axis with X-oriented polarization in agreement with the experimental conditions. For a single-void geometry, the calculations are realized for the void of a specific size in a homogeneous environment or on a semi-infinite substrate surrounded by perfectly matched layers mimicking an infinite region.

The quality factor of modes is calculated from the complex resonance frequency as its real part divided by twice the magnitude of its imaginary part. To take into account the material dispersion, we apply the following iteration procedure. We estimate the mode wavelength analytically, evaluate the refractive index for the guess wavelength and search for complex eigenfrequencies with this fixed refractive index. Then, we calculate the resonant wavelength from the resulting complex eigenfrequency and use it as the input for refractive index. We repeat the procedure iteratively until it converges. Our criterion of convergence is the relative difference between input and output wavelengths of <0.01%. We carry out the iterative search for each mode (ED, MD, EQ, MQ) independently. To plot Fig. 2c, we determine resonant wavelengths and quality factors sweeping the resonator radius. While sweeping the resonator radius, we use the calculated mode wavelengths for some radius as the initial guess for the resonator with a slightly larger or smaller radius. Next, we establish one-to-one correspondence between wavelengths and quality factors and obtain the dependence of quality factor on the resonant wavelength.

Scattering Problem of Periodic Arrays. The simulations for the periodic arrays (Fig. 3 and Supplementary Figs. S6 and S8) were performed using the frequency-domain solver of the commercial finite-element package COMSOL Multiphysics. The voids were modelled as conical holes in an infinitely-high silicon substrate (refractive index taken as in the top panel of Fig. 2c) and with an infinitely-high layer of air on top (refractive index 1). The infinite extension was mimicked in the finite-element method with perfectly matched layers at the top and bottom of the calculation domain. The hole dimensions were estimated from scanning electron microscope images of the fabricated samples and are provided under Fig. S6. In accordance with the fabricated structures, the system was assumed to be periodic in lateral directions with a period of 900 nm. This was accounted for via periodic boundary conditions. For the excitation, a periodic port was placed in the air domain above the voids. In consistency with the experimental setup, the system was excited under normal incidence and the reflectivity was defined as the power that is back-reflected into the zeroth diffraction order divided by the input power. The electric fields in Fig. 3 and
Supplementary Fig. S8 were extracted at the wavelengths of 439 nm and 709 nm, close to the peaks of the reflectivity spectrum for the depicted void dimensions. In all simulations, it was ensured that the mesh resolution is chosen high enough so that convergence is reached. Furthermore, the spectra calculations have been cross-checked by an in-house implementation of the Fourier-modal method\(^{45,66}\). This method was also used to calculate the large diameter and depth sweeps in Fig. 4. For the sidewalls of the voids, a constant inclination angle of 15° is assumed, which is discretized in the calculations by a staircasing that is finer than the spatial accuracy of the underlying plane-wave expansion, the parameters for depth and diameter can be found in Tables ST1 and ST2 in the Supporting Information.

**Multipolar Decomposition.** For characterization of the eigenmode spectrum in Fig. S8, we apply the decomposition method over irreducible spherical multipoles, characterized by both azimuthal \((m)\) and orbital \((l)\) indices. The azimuthal index is fixed to \(|m| = 1\) to ensure that the modes are excitable with a linearly polarized plane wave. The multipolar decomposition is realized as custom built-in routines for the eigenmode solver in COMSOL Multiphysics. For multipolar channel classification, we use electric and magnetic dipolar harmonics \((l = 1)\), electric and magnetic quadrupolar harmonics \((l = 2)\), electric and magnetic octupolar harmonics \((l = 3)\), and electric and magnetic hexadecapolar harmonics \((l = 4)\). We note that the eigenmodes of particles and voids of spherical geometry can be classified unambiguously according to the multipolar structure of their far-field radiation, thus, we label them as electric (ED) and magnetic (MD) dipole, and electric (EQ) and magnetic (MQ) quadrupole. For non-spherical geometries such correspondence holds only approximately.

**Data and material availability**

All data needed to evaluate the conclusions in the paper are present in the paper and/or the Supplementary Materials. Additional data related to this paper may be requested from the authors.

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**Author contributions**

M.H. conceived the original idea. All authors discussed, interpreted, and corroborated the experimental results, simulation, and interpretation. M.H., T.W., Y.K., and H.G. supervised the work. K.K., S.B., L.S., and T.W. performed the simulations and modelling. M.H. fabricated and characterized the samples. F.S. performed the spectroscopic measurements. J.K. and M.H. designed, manufactured, and measured the colour samples and art work. All authors participated in the preparation and writing of the manuscript.

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**Conflict of interest**

The authors declare no competing interests.

**Supplementary information**

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