Structural and optical properties Investigation of Zn_xCd1-xS thin films

Nada K. Abbas¹, Ahlam M. Farhan², Naan F. Majeed¹ and Suaad A-Muhammed¹
1Department of Physics College of Science for Women, Baghdad University, Baghdad. Iraq
E-mail: nadabbs@yahoo.com
2 Department of Chemistry College of Science for Women, Baghdad University, Baghdad. Iraq
ahlam63a@yahoo.com
1Department of Physics College of Science for Women, Baghdad University, Baghdad. Iraq
neean89@yahoo.com

Abstract – CdZn_xS_1-x thin films with different composition have been deposited on glass substrate by the spray pyrolysis method at RT using CdCl₂ (0.1M), ZnCl₂(0.1M) and H₂NCSNH₂(0.1M)solution and a substrate temperature of (400±20°C). X-ray diffraction studies reveal that the films are polycrystalline in nature with hexagonal structure and preferential orientation along (002). The grain size of the films is found to increase from (37.397 to 46.902) nm with increasing Zinc concentration while the strain and the dislocation density of the films are found to decrease from (7.15 to 4.54) × 10⁴ rad and from (3.82 to 1.93) × 10¹⁴ lines.m⁻² respectively. The transmittance spectrums of CdZn_xS_1-x thin films reveal very pronounced interference effects for photon energies below the fundamental absorption edge by exhibiting interference pattern. The optical energy gap for CdZn_xS_1-x thin films increases and shifts towards the UV region as the Zn concentration in the films increased.

Keywords: CdZn_xS_1-x thin films, CdS thin films, ZnS thin films, spray pyrolysis technique

Council for Innovative Research
Peer Review Research Publishing System

Journal: Journal of Advances in Chemistry
Vol. 12, No. 3
www.cirworld.com, editor@cirworld.com
Introduction

Recent investigations have evoked considerable interest in ZnS thin films due to their vast potential for use in thin film devices such as photo luminescent and electroluminescent devices and more recently as n-type window layer heterojunction solar cells [1]. Zinc sulfide has found wide use as a thin film coating in the optical and microelectronic industries. It has high refractive index (2.2±0.5 at 632 nm), high effective dielectric constant (9 at 1 MHz) and wide wavelength pass band (0.4–13 mm) [2]. It is commonly used as filter, reflector and planar waveguide.

Cadmium sulphide (CdS) are considered to be very important materials for a wide spectrum of optoelectronic applications as having good chemical, mechanical stability [3] and specific physical properties such as direct band gap widths, high absorption coefficients in the visible and infrared part of the solar spectrum, good electrical properties (e.g. carrier mobility and lifetime) and increased capability in obtaining adjustable n- or p-type conductivity by doping [4].

Thin film of Zn₁₋ₓCdₓS are known to have properties in between those of CdS and ZnS. Because the addition of Zinc to cadmium sulphide has resulted in very interesting properties related to photo electrochemistry and optoelectronics [5]. Because the ternary materials provide a possibility of tailoring their properties as per requirements and hence project themselves as important semiconducting materials for the applications in the field of device fabrication [6]. The band gap energy of CdS can be increased by the addition of ZnS (Eg=3.6 eV). The ternary compound Zn₁₋ₓCdₓS offers a great range of tune ability both of its band gap (from 2.42 eV for CdS to 3.6 eV for ZnS) and its lattice parameters [7].

Several techniques were employed for the growth of the ternary CdZnS films. In the present work, Zn₁₋ₓCdₓS thin films were deposited by Spray pyrolysis technique at different composition of (x). The effect of Zinc concentration on the structural and optical properties of these films was studied and discussed.

2. Experimental Details

ZnₓCd₁₋ₓS thin films were produced on a glass substrate by the spray pyrolysis technique. The ZnCl₂, CdCl₂ salts and H₂NCSNH₂ were dissolved in deionized water in separate beakers. Aqueous solutions of ZnCl₂, CdCl₂ salts and H₂NCSNH₂ were used as the sources of Zn, Cd, S, respectively. The ZnCl₂, CdCl₂ and H₂NCSNH₂ solutions were mixed for 30 min. With a magnetic stirrer. The compositions of the solutions used to fabricate the ZnₓCd₁₋ₓS thin films are shown in Table 1 in terms of the nominal concentrations in the deposition solution. The substrate temperature was regulated at (400±20°C) during the deposition process using a resistive heater and a thermocouple. Glass substrates were prepared by cutting (2.5×2.5 cm²) pieces and cleaning them by water and they were placed in the microwave until we used them.

In order to spray the solution onto the substrate using an ultrasonic atomizer, nitrogen (N2) was used as the carrier gas at a pressure of 3 bar with a deposition rate of 3 cm/min during the deposition process. At the end of the spraying process, nitrogen (N2) was flowed onto the thin films formed on the glass substrate for 3 min in order to dry them. They were then cooled down naturally to room temperature.

Flowed onto the thin films formed on the glass substrate for 3 min in order to dry them. Films thickness was determined by a multiple beam interferometry (Fizeau fringes in reflection). The films were characterized by X-ray diffraction technique using Philips X-ray diffractometer with Cu-Kα radiation with wavelength (1.5406 Å). A UV-visible spectrophotometer supplied by Japanese company (Shimadzu) was used to record the optical absorbance and transmittance spectra of ZnₓCd₁₋ₓS thin films at wavelength range (400-1100 nm).

3. Result and discussion:

3.1. Structural properties:

The X-ray diffraction patterns of ZnₓCd₁₋ₓS thin films where x equal (0, 0.25, 0.5, 0.75& 1) deposited by spray pyrolysis method on glass substrate at R.T with thickness (500 nm) are shown in Fig. (1). The XRD patterns for all cases reveal polycrystalline in nature for as-deposited films having the main diffraction peak corresponding to the reflection from (002) plane. The diffraction peaks for (002) are located at (27.07, 27.37, 27.76, 28.06, 28.55) for x equal (0, 0.25, 0.5, 0.75, 1) respectively. Therefore, it is clear that there is a shift toward higher value of 2θ when composition (x) change from 0 to 1.

As compared with ASTM cards, all films exhibit pure hexagonal structure and our results agree with [8]-[15]. The X-ray peak corresponding to (002) reflection is observed in all cases which represents the
The preferential orientation in Zn$_x$Cd$_{1-x}$S films. The intense and sharp peaks in the XRD pattern reveal the good crystallinity of the thin films and confirm the stoichiometric nature of Zn$_x$Cd$_{1-x}$S thin films.

The preferential orientation (002) in Zn$_x$Cd$_{1-x}$S thin films where x= 0 (CdS film) it was reported for thermal vacuum evaporated [16] - [17] - [18] - [19] while it was reported for chemical bathed[20]-[21]. Whereas for x equal to 0.5 and for x=0.7 has been also reported for chemical deposited [22] [23]. Finally for x= 1 (ZnS film) the preferential orientation (002) was reported for development technology solution [24] while it was reported for Chemical decomposition[25]. Table (2) illustrates bragg's angles , inter planers spacing , relative intensities , miller indices , and lattice constants of Zn$_x$Cd$_{1-x}$S thin films.

The diffraction peaks for (002) are located at 27.07°, 27.37°, 27.76°,28.06° and 28.55° for x equal ( 0 , 0.25, 0.5 , 0.75 & 1 ) respectively , therefore it is clear that there is a shift toward higher value of 2θ when composition (x) change from 0 to 1. This shifting has been reported Zia [14]. This shifting in the peak position with increasing composition (x) suggests that the lattice constants increase with the increasing in Zn concentration as listed in table (2) and calculated from the following equation [26]

\[
\frac{1}{d^2} = \frac{4}{\lambda^2} \left( \frac{h^2 + h^2 + k^2}{c^2} \right) + \frac{l^2}{a^2} \tag{1}
\]

Where \(d\) : is the interplaner distance., \(hkl\) : miller indices. \(a,c\) : lattice constants.

The calculated values of lattice constants for in Zn$_x$Cd$_{1-x}$S thin films are in good agreement with ASTM data. Similar results have been reported by Ghoneim [13].

It is clear from XRD patterns of in Zn$_x$Cd$_{1-x}$S thin films , that the full width at half maximum (FWHM) decreases with the increasing of Zn concentration in these films. This decreasing in (FWHM) indicates an increasing in the grain size of in Zn$_x$Cd$_{1-x}$S thin films as given in table (3) according to Scherrer's formula where the relation between the grain size (D) and (FWHM) is reversal as follows [27] :

\[
D = \frac{0.9 \lambda}{\beta \cos \theta} \tag{2}
\]

Where \(\beta\) is the full width at half maximum (FWHM) in radian and \(\lambda\) is the X-ray wavelength (1.5406 Å).

The increasing in the grain size with increasing in the Zn concentration had been also reported for chemical path deposition [28]-[21].

![Fig1 X-ray diffractograms of thin film for different concentration](image)

---

**ISSN 2321-807X**

- 4267 - Page 2

January 25, 2016
Table (2) the structural parameters of Zn$_x$Cd$_{1-x}$S thin films

| thin film     | 2 $\Phi$ | d(exp) | d(ASTM) | Hkl | a(nm) |
|---------------|----------|--------|---------|-----|-------|
| CdS           | 25       | 0.3565 | 0.358   | 100 | 0.4116|
|               | 27.0751  | 0.329  | 0.337   | 002 | -     |
|               | 28.753   | 0.310  | 0.316   | 101 | 0.405 |
|               | 37.2412  | 0.241  | 0.245   | 102 | 0.408 |
|               | 43.8541  | 0.206  | 0.2068  | 110 | 0.409 |
|               | 48.3943  | 0.1879 | 0.1898  | 103 | 0.42  |
| Zn$_{0.25}$Cd$_{1-0.25}$S | 25.1011 | 0.3543 | 0.3542  | 100 | 0.375 |
|               | 27.3712  | 0.325  | 0.325   | 002 | -     |
|               | 29.1478  | 0.3056 | 0.3054  | 101 | 0.399 |
|               | 37.4386  | 0.239  | 0.235   | 102 | 0.407 |
|               | 43.8541  | 0.2062 | 0.2059  | 110 | 0.412 |
|               | 54.2306  | 0.17187| 0.1718  | 200 | 0.3969|
| Zn$_{0.5}$Cd$_{1-0.5}$S  | 25.198   | 0.353  |         | 100 | 0.407 |
|               | 27.766   | 0.321  | 002     | -   |       |
|               | 29.2465  | 0.305  | 101     | 0.4 |       |
|               | 37.8334  | 0.2375 | 102     | 0.407|     |
|               | 43.5581  | 0.2069 | 110     | 0.413|     |
|               | 49.0852  | 0.1854 | 103     | 0.48 |     |
|               | 54.2176  | 0.1689 | 200     | 0.39 |     |
| Zn$_{0.75}$Cd$_{1-0.75}$S | 25.6933 | 0.346  | 100     | 0.404|     |
|               | 28.0621  | 0.3176 | 002     | -   |       |
|               | 29.3452  | 0.304  | 101     | 0.4015|    |
|               | 37.8334  | 0.2375 | 102     | 0.413|    |
|               | 44.463   | 0.2037 | 110     | 0.407|    |
|               | 49.0852  | 0.1854 | 103     | 0.443|    |
|               | 55.3033  | 0.166  | 200     | 0.383|    |
| ZnS           | 26.8777  | 0.3313 | 0.3309  | 100 | 0.382 |
|               | 28.5556  | 0.3122 | 0.3128  | 002 | -     |
|               | 30.4309  | 0.2934 | 0.2925  | 101 | 0.383 |
|               | 39.61    | 0.2272 | 0.2273  | 102 | 0.3825|
|               | 47.4073  | 0.1915 | 0.1911  | 110 | 0.383 |
|               | 51.8488  | 0.1763 | 0.1764  | 103 | 0.3829|
|               | 55.3033  | 0.1659 | 0.1661  | 200 | 0.383 |

The dislocation density ($\delta$) of in Zn$_x$Cd$_{1-x}$S thin films, which defined as the length of dislocation lines per unit volume of the crystal, was calculated from this equation [29]:

\[ \delta = \frac{1}{V} \frac{1}{L} \]
\[ \delta = \frac{1}{d^2} \]  
\( (3) \)

The values of the dislocation density of in Zn\(_x\)Cd\(_{1-x}\)S thin films are given in table (3). It is evident from this table that the dislocation density of in Zn\(_x\)Cd\(_{1-x}\)S thin films decreases with increasing in Zn concentration which can be also deduced from the increasing in the grain size where the dislocation density is proportion reversely with the square of the grain size according to eq.(3). This decreasing in the dislocation density indicates an improvement in the crystallinity of in Zn\(_x\)Cd\(_{1-x}\)S thin films and their homogeneity as increasing in Zn concentration.

The strain (\(\xi\)) developed in Zn\(_x\)Cd\(_{1-x}\)S thin films can be calculated from the relation [30]:

\[ \xi = \frac{\beta \cot \beta}{4} \]  
\( (4) \)

The origin of strain is related to lattice misfit which in turn depends upon the growing condition of the films [31]. The values of the strain of in Zn\(_x\)Cd\(_{1-x}\)S thin films are given in table (3). It is clear from this table that the strain in Zn\(_x\)Cd\(_{1-x}\)S thin films decreases with increasing in Zn concentration which can be also deduced from the increasing in the grain size, where the decreasing in the strain and the dislocation density with the increasing in the grain size is a well-known phenomenon [32]. In polycrystalline films, the dislocated atoms occupy the regions near the grain boundary. Due to large number of grain boundaries and short distance between them, the intrinsic strains are always associated with such interface. The increasing in the grain size causing reduction in the number of grain boundaries and that leads a reduction in the intrinsic strains associated with the grain boundaries interface.

Table 3: variation of the full width at half maximum, grain size, dislocation density and no.of crystals of Zn\(_x\)Cd\(_{1-x}\)S thin films with composition(x) .

| Thin film          | Grain size (nm) | \(\delta\) (line/m²)¹ | N₀ -10³m⁻² |
|--------------------|-----------------|-------------------------|------------|
| CdS                | 37.397          | 7.15                    | 3.82       |
| Zn\(_{0.25}\)Cd\(_{0.75}\)S | 37.13          | 7.25                    | 3.9        |
| Zn\(_{0.5}\)Cd\(_{0.5}\)S   | 47.02           | 4.52                    | 1.92       |
| Zn\(_{0.75}\)Cd\(_{0.25}\)S | 46.805         | 4.56                    | 1.9        |
| ZnS                | 46.902          | 4.54                    | 1.93       |

**Optical properties**

The transmittance spectrum of Zn\(_x\)Cd\(_{1-x}\)S thin films where x equal (0, 0.25, 0.5, 0.75, 1) are shown in Fig.(2). It is clear from the figure that the transmittance increases with increasing in Zn concentration which can be also deduced from the changing in the films' colour. This changing in the films' colour was also mentioned by Chaudhari et al. [33]. Also our results agree with Nadeem et al. [34] who found that with increasing in Zn concentration from 0 to 1 the transmittance of Zn\(_x\)Cd\(_{1-x}\)S thin films increases. Also it is clear that the transmittance of Zn\(_x\)Cd\(_{1-x}\)S thin films changes from \(\approx 70\%\) to \(\approx 90\%\) and this is consider a wide range which can be useful in different application like optical filters and that agree with Kumar[20]. In addition the variation of the transmittance of Zn\(_x\)Cd\(_{1-x}\)S thin films with the wavelength is very important because this variation will limit the transmitted wavelengths which play an important role in determination the category/type of the optical filters. The absorbance spectrums of Zn\(_x\)Cd\(_{1-x}\)S thin films where x equal (0, 0.25, 0.5, 0.75, & 1) are shown in Fig. (3). It is clear that as the Zn concentration increases the absorbance of Zn\(_x\)Cd\(_{1-x}\)S thin films is decreased. This decreasing in the absorbance is attributed to the decreasing of Cd concentration which results in an decrease of the depth of donor levels and these levels will be available for the photons to be absorbed therefore the absorbance of Zn\(_x\)Cd\(_{1-x}\)S thin films will decrease with increasing in Zn concentration. As well as from the same figure, it can be seen that the absorption edge shifts to the lower wavelengths as the Zn concentration increased and takes the values 506 nm, 496 nm, 458 nm, 386 nm and 364 nm for x equal 0, 0.25, 0.5, 0.75 and 1 respectively. This shifting in the absorption edge was also mentioned by [15]-[23]. From this shifting in the absorption edge it can be deduced that the energy gap of Zn\(_x\)Cd\(_{1-x}\)S thin films will increases with increasing Zn concentrations.
The optical energy gap values \( E_g \) for CdSe, \( x \)-Cd, thin films prepared by thermal evaporation method have been determined from the region of the high absorption at the fundamental absorption edge of these films by using Tauc equation [35]:

\[
\alpha \cdot h \nu = B_0 \left( h \nu - E_g \right)^n \tag{5}
\]

Where \( \alpha \) is the absorption coefficient, \( h \nu \) is the incident photon energy in eV, \( B_0 \) is a constant depends on the nature of the material (properties of its valence and conduction band) [36], and \( n \) is a constant depends on the nature of the transition between the top of the valence band and bottom of the conduction band [37].

This equation is used to find the type of the optical transition by plotting the relations \((\alpha h \nu)^2\), \((\alpha h \nu)^{2/3}\), \((\alpha h \nu)^{1/2}\), and \((\alpha h \nu)^{1/3}\) versus photon energy \((h \nu)\) and select the optimum linear part. It is found that the first relation yields linear dependence, which describes the allowed direct transition, then \( E_g \) was determined by the extrapolation of the portion at \( \alpha = 0 \) as shown in Fig.(4). It is clear that the optical energy gap for \( \text{Zn}_x \text{Cd}_{1-x} \text{S} \) thin films increases as the \( \text{Zn} \) concentration in the films increased. This is attributed to the decreasing of \( \text{Cd} \) concentration which results increase of the depth of donor levels associated which in turn causing a increasing the optical energy gap for \( \text{Zn}_x \text{Cd}_{1-x} \text{S} \) thin films.

For \( x = 1 \) and \( x = 0 \), the energy gap is determined by the extrapolation of the portion at \( \alpha = 0 \) as shown in Fig.(4). It is clear that the optical energy gap for \( \text{Zn}_x \text{Cd}_{1-x} \text{S} \) thin films increases as the \( \text{Zn} \) concentration in the films increased. This is attributed to the decreasing of \( \text{Cd} \) concentration which results increase of the depth of donor levels associated which in turn causing a increasing the optical energy gap for \( \text{Zn}_x \text{Cd}_{1-x} \text{S} \) thin films.

The optical energy gap values \( E_g \) for \( \text{CdSe}, x = 0 \) thin films have been also reported for Martin et.al. [23]. The optical energy gap values for \( \text{Zn}_x \text{Cd}_{1-x} \text{S} \) thin films were 2.45 eV, 2.5 eV, 2.7 eV, 3.2 and 3.43 eV for \( x \) equal (0, 0.25, 0.5, 0.75 & 1) respectively as shown in table 4. The obtained values of the optical energy gap match well with the reported values of CdS [38]-[39][40] and Zns [34]-[41], the other composition have in between values.
Table 4. Comparison of band gap, energy wavelength of ZnxCd1-xS film

| X=   | Eg(ev) | \(\lambda\)(nm) |
|------|--------|------------------|
| 0    | 2.4    | 516              |
| 0.25 | 2.5    | 496              |
| 0.5  | 2.8    | 443              |
| 0.75 | 3.15   | 394              |
| 0.75 | 3.15   | 394              |

II. Conclusion

Chemical spray pyrolysis technique can be successfully employed for the deposition of uniform morphologist and polycrystalline ZnxCd1-xS thin films with hexagonal phase at R.T. The increasing of Zn concentration in these films has improved the crystallinity of the films and their homogeneity because the decreasing in the strain and the dislocation density with the increasing in the grain size according to the increasing of Se concentration. Morphological studied indicates that surface roughness decreases with increasing in Zn concentration. Optical studied indicates that ZnxCd1-xS thin films exhibit direct band gap which is strongly depends on the Zn concentration almost cover the entire visible spectral that makes these films are suitable for optoelectronic devices especially for solar cell and optical filter.

Reference

[1] Ortega Borges R, Lincot D and Vedel J 1992 Paper presented at11th European photovoltaic solar energy conference, Montreux

[2] Orient Tom 1994 J. Electrochem. Soc. 141 1320.

[3] Singh R. S., and Bhushan S., 2010, Structural and optical studies of chemically deposited CdS-Se films , Journal of Non-Oxide Glasses Vol. 2, No 3, p. 135 – 141.

[4] Singh R. S. and Bhushan S., 2009, Structural and optical studies of chemically deposited Cd(S-Se):CdCl2 Sm films , Bull Mater. Sci Vol. 32, No 2, p. 125 – 133.

[5] Murali, K. R., Elango, P., Andavan, P., and K. Venkatachalam, 2008, Preparation of CdS, Se-x films by brush plating technique and their characteristics , Journal of Materials Science: Materials in Electronics, Vol. 19, No. 3, p. 289-293.

[6] Chate, P. A., Sathe, D. J., and Hankare, P. P., 2011, Electrical and crystallographic properties of nanocrystalline CdSe0.5S0.5 composite thin films deposited by dip method , Journal of Materials Science: Materials in Electronics, Vol. 22, No. 2, p. 111-115.

[7] T.D.Dzhafarov, Formation of CdZnS thin film by Zn diffusion, Yildiz Technical University Istanbul, Turkey(2010).

[8] Makads,M. Nada, K., Lamia K. 2011, Optical Investigations of CdSe1-x Tex Thin Films, Baghdad Science Journal, Vol.8(1),pp:123-133.

[9] Metin Bedir, Refik Kayali and Mustafa Oztas “Effect of the Zn concentration on the characteristic parameters (Zn,Cd,S) films developed by spraying phyrolysis method under the Nitrogen Atmosphere” Türk J Phys 26(2002) pp.121-126.

[10] T.Prem Kumar ,K. Sankaranarayanan “Tunability of structural,surface texture, compositional and optical properties of CdZnS thin film by photo assisted chemical bath deposition technique” Chalogenide Letters, Vol. 6, No. 11(2009) p617-622.

[11] Ayush Khare “Effects of the Zn concentration on Electro-optical properties of Zn,Cd,S films” Chalogenide Letters Vol.6,N0.12(2010) P.661-671.

[12] Asogwa, P.U..2010, “Variation of optical properties with post deposition annealing chemically deposited CdZnS thin films” Chalogenide Letters Vol.7,No.8, p501-508.
[13] D. Ghoneim "Influence of zinc on optical, electrical and structural properties of (Zn,Cd₅₋ₓ)S films" Chalcogenide Letters Vol. 7, No. 6 (2010) p. 413-422.

[14] Rehana Zia "Preparation and Characterization of Thin Film Polycrystalline Solar Cells" Doctors thesis, Department of Physics, University Lahore college for women (2008).

[15] Maysa Tayseer "Preparation and Enhancement of CdS/ZnS thin film for photovoltaic purposes" Thesis of Master/An-Najah National University, Nablus (2010).

[16] Zhibing He, Gaoling Zhao, Wenjian Weng, Piyi Du, Ge Shen and Gaorong Han, The difference between the transverse and in-plane resistivity of vacuum evaporated cadmium sulfide (CdS) thin films, Vacuum, Vol. 79, Issues 1-2, 8 July (2005), p. 14-18.

[17] A. Ashour, The physical characteristics of Cu₂S/CdS thin film solar cell, Journal of Optoelectronics and Advanced Materials Vol. 8, No. 4, August (2006), p. 1447 – 1451.

[18] Ion, L. Enculescu, I. Iftimie, S., Ghenesuc, V., Tazlaoanu, C. Besleaga, Mitran, T. L., Antohe, V. A. Gugiu and M. M. Antohe, S., 2010, Effect of proton irradiation on the spectral performance of photovoltaic cells based on CdS/CdTe thin films, Chalcogenide Letters Vol. 7, No. 8, p. 521-530.

[19] Z.R. Khan, M. Zulfiquar and M.S. Khan, Effect of thickness on structural and optical properties of thermally evaporated cadmium sulfide polycrystalline thin films, Chalcogenide Letters Vol. 7, No. 6, June (2010), p. 431-438.

[20] T. Prem Kumar, S. Saravana Kumar and K. Sankaranarayanan, "Effect of annealing on the surface and band gap alignment of (CdZnS) thin films" Applied Surface Science, 255 (2010) 235–241.

[21] Zoubeida Khefachaa, Zohra Benzartib and Mohamed Mnaria, "Electrical and optical properties of Cd₁₋ₓZnₓS (0<x<0.18) grown by chemical bath deposition" Journal of Crystal Growth 260 (2004) p400-409.

[22] T. Prem Kumar, K. Sankaranarayanan, "Tunability of structural, surface texture, compositional and optical properties of (CdZnS) thin film by photo assisted chemical bath deposition technique" Chalcogenide Letters, Vol. 6, No. 11 (2009) p617-622.

[23] Metin Bedir, Refik Kayali and Mustafa Oztas, 2002, "Effect of the Zn concentration on the characteristic parameters of (Zn,Cd₅₋ₓ)S films developed by spray pyrolysis method under the Nitrogen Atmosphere" Turk J Phys 26, pp.121-126.

[24] G. K. Padam, G. L. Malhotra and S. U. M. Rao, Phys. V. 63, No.3, PP. 770-774.

[25] M. Celaeltin Baykul, Nilgun Orhan, 2010, "Band alignment of (Cd₁₋ₓZnₓS) produced by spray pyrolysis method" Thin Solid Films 518 p1925-1928.

[26] K. Sivaramamoorthy, S. Asath Bahadur, M. Kottaisamy and K. R. Murali, 2010, Structural, optical and photoconductive properties of electron beam evaporated CdS,Se, alloy films, Crystal Research and Technology Vol. 45, Issue 4, pp. 414-420.

[27] Y. Sitrotin and M. Shaskolskaya, "Fundamental of crystal physics", (Mir publishers, Moscow, 1982).

[28] M. Dhanam, B. Kavitha "Influence of tea (complexing agent) on the structural properties of CBD ZnS Thin Films" Chalcogenide Letters Vol. 6, No. 7 (2009) P 299-307.

[29] M. M. EL Nahass and A. M. A. EL Barry, "Effect of substrate temperature, deposition rate and heat treatment on structural and carrier transport mechanisms of thermal evaporated p-Cd₁₋ₓS/n-CdS heterojunction", Indian Journal of Pure & Applied Physics Vol. 45, May (2007), p. 485-487.

[30] Z.R. Khan, M. Zulfiquar and M.S. Khan, "Effect of thickness on structural and optical properties of thermally evaporated cadmium sulfide polycrystalline thin films", Chalcogenide Letters Vol. 7, No. 6, June (2010), p. 431-438.

[31] K. Sarmah, R. Sarma and H. L. Das, "Correlative assessment of structural and photoelectrical properties of thermally evaporated CdSe thin films", Journal of Non-Oxide Glasses Vol. 1, No. 2, (2009), p. 143-156.

[32] Suthan Kissinger, N.J., Suthagar, J., Saravana Kumar, B. Balasubramaniam T. and Perumal K., 2010, "Effect of substrate temperature on the structural and optical properties of nanocrystalline cadmium selenide thin films prepared by electron beam evaporation technique", ACTA PHYSICA POLONICA A, Vol. 118, No. 4, p. 623-628.

[33] Chaudhari, J.B., Deshpande, N.G., Gudage, Y.G., Ghosh, Huse and Ramphal Sharma V.B., 2008, "Studies on growth and characterization of ternary CdS₁₋ₓSe, alloy thin films deposited by chemical bath deposition technique", Applied Surface Science, Vol. 254, p. 6810-6816.

[34] Nadeem, M.Y., waqas, A., 2000, "Optical Properties of ZnS Thin Films" Turk. j. Phy. V.24, pp. 651-659.

[35] Mott N. F., and Davis, E. A., 1979, "Electronic Processes in Non-Crystalline Materials" , 2nd, Clarendon Press, Oxford.

[36] Kasap, S. O., 2002, "Principle of electronic materials and devices", (Sneded, McGraw-Hill, New York).
[37] Z.R. Khan , M. Zulfequar and M.S. Khan , Effect of thickness on structural and optical properties of thermally evaporated cadmium sulfide polycrystalline thin films , Chalcogenide Letters Vol. 7, No. 6, June (2010) , p. 431-438 .

[38]. Yadav, A.A. and Masumdar , E.U.,2010,Optical and electrical transport properties of spray deposited CdS1−xSex thin films , Journal of Alloys and Compounds Vol. 505 , p. 787-792 .

[39]. ] Rusu, M. Rumberg, A. Schuler, S. Nishiwaki, R. Wu’rz, S.M. Babu, M. Dziedzina, KelchC., S. Siebentritt, R. Klenk, Th. Schedel-Niedrig, M.Ch. Lux-Steiner ,2003, Optimization of the CBD CdS deposition parameters for ZnO/CdS/CuGaSe2/Mo solar cells , Journal of Physics and Chemistry of Solids Vol. 64 , (2003), p.1849–1853.

[40].Abdullah Serhan AL-Shammari,2005, "Preparation and characterization of chlorine doped cadmium sulphide (CdS,CL)thin film and their application in solar cells "Thesis of Master, King Saud University, college of Science.

[41]. Dedova,T., Krunks, M., Volobujeva and I. O.,2004, Department of Materials Science, "Tallinn University of Technology", Estonia.

Authors’ information

Professor Nada Khdair Abbas, Ph.D. in the Department of Physics, College of Science for Women, University of Baghdad .She had awarded the B.Sc. degree from the department of physics in 1986, M.Sc. in 1995 and Ph.D. in 2000. Teaching Students of under and post graduate, Chief of solid and materials group, Chairman department of physics. Committee Chairman promotions in the college of Science for women. She interested with research about Nanotechnology, the thin films and their use in solar cells and detectors.

Prof. Dr. Ahlam M. Farhan : Acting Dean of College of Science for Women Ph.D. in the Department of Chemistry, College of Science for Women, University of Baghdad She had awarded the B.Sc. degree of from the department of Chemistry in 1985, M.Sc. in 1995 and Ph.D. in 2000. Teaching Students of under and post graduate.

Neean F.Mageed: was born in Baghdad in 1989 .She had awarded the B.Sc. degree from the department of physics, College of Science for Women, University of Baghdad in the year 2009/2010. She works as assistant of physicist in college of science for women, physics department, Baghdad University. She interest in field of solid materials and nanotechnology research