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Crystalline boron nitride nanosheets by sonication-assisted hydrothermal exfoliation

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Abstract: A simple method to prepare two-dimensional hexagonal boron nitride (h-BN) scalably is essential for practical applications. Despite intense research in this area, high-yield production of two-dimensional h-BN with large size and high crystallinity is still a key challenge. In the present work, we propose a simple exfoliation process for boron nitride nanosheets (BNNSs) with high crystallinity by sonication-assisted hydrothermal method, via the synergistic effect of the high pressure, and cavitation of the sonication. Compared with the method only by sonication, the sonication-assisted hydrothermal method can get the fewer-layer BNNSs with high crystallinity. Meanwhile, it can reach higher yield of nearly 1.68%, as the hydrothermal method with the yield of only 0.12%. The simple sonication-assisted hydrothermal method has potential applications in exfoliating other layered materials, thus opening new ways to produce other layered materials in high yield and high crystallinity.

Keywords: boron nitride nanosheet (BNNS); high crystallinity; sonication-assisted; hydrothermal method

1 Introduction

In recent years, the discovery of graphene, a one-atom thick flat allotrope of carbon, has inspired researchers to explore other two-dimensional materials for the unique properties and various promising applications [1–4]. Among the two-dimensional materials, atomically thin hexagonal boron nitride (h-BN), has attracted more attention due to the excellent chemical stability, outstanding mechanical and thermal properties [5], and potential applications in ultraviolet-light emitters, high-temperature oxidation-resistant coatings, and an ideal substrate for high-quality graphene electronics [6–11]. All of these applications need an amount of high-crystallinity BNNSs. Although the BNNS is the structure analogue of graphene, some methods usually used for synthesizing graphene, such as the Hummer’s and related graphene oxide methods, are not suitable for the preparation of BNNS, because of slightly ionic bonding between the BN layers (i.e., lip–lip interactions), which is stronger than the van der Waals’ force between graphite layers.

With years’ efforts, several modified methods for effectively fabricating few-layer BNNSs have been...
Experimental procedure

It should be pointed out that it is impossible to make sure every batch of commercially-obtained h-BN powders is identical. Notably, the chemistry properties of h-BN are very dependent on BN sample preparation, purity, crystallinity, and microstructure. Because of this, the results of the experiment may be different from batch to batch.

In order to get BNNSs in sonication-assisted hydrothermal conditions, the h-BN powders (Alfa Aesar Co.) and 60% tertiary butanol solution (Beijing Chemical Co., Beijing, China) were fed into a poly(tetrafluoroethylene)-lined stainless steel. Then, the reactor was placed into the sonication cleaner with the rated power of 300 W, and kept in the temperature of 355 K for 2 h, as the schematic shown in Fig. 1. After sonication-assisted hydrothermal exfoliation, the dispersions were transferred into a tube and centrifuged at 3500 rpm for 30 min. Then, the top 90 mL supernatant (out of 100 mL) was carefully collected to characterize the BNNSs, and the sediment could be cycled for further hydrothermal exfoliation. With the purpose of proving the valid of the method, the controlled trials were designed and conducted. In order to distinguish these methods, the h-BN powders treated by sonication were named as SC-BN, treated by hydrothermal named as HT-BN, and treated by sonication-assisted hydrothermal named as SH-BN, respectively.

The microstructure of raw BN powders and sediment of centrifugal treatment were observed using a scanning electron microscope (SEM; Melin, Zeiss Corporation, Germany) at an accelerating voltage of 5.0 kV. The morphology and the diffraction pattern of
BNNS were detected by high-resolution transmission electron microscope (TEM; JEM-2100F, JEOL, Japan), operated at an accelerating voltage of 200 kV. The TEM samples were prepared by drying a droplet of the BNNS suspension on a carbon grid. The thickness and the lateral size of the BNNS were analyzed by the atomic force microscopy (AFM) images obtained with a Nanoscope MultiMode instrument (Digital Instruments/Bruker Systems), operated in the air tapping mode. The samples were prepared by depositing the BNNS suspension on a Si substrate and dried in a vacuum oven for 2 h before AFM measurement. Raman spectra were excited using the 633 nm excitation source from He–Ne laser and collected by a micro-Raman spectrometer (HR800, HORIBA, France) at room temperature.

Transmission electron microscopy (TEM) was employed to count the thickness and further analyze the structures of the BNNS exfoliated by the sonication-assisted hydrothermal method. Some of the stacked BNNSs with single or a few layers in larger size can be observed (Fig. 3(a)). Due to ultimately thin shapes, the BNNSs were entirely transparent to an electron beam, making it possible to count the layers. As shown in Fig. 3(b), it is obvious that the suspended films are consisted of the few-layer BNNSs (4 to 6 layers). The incident electron beam is along the [002] direction, thus perpendicular to the (002) crystal face. Through the electron diffraction pattern of the BNNS, the pattern reveals a typical six-fold symmetry natural of h-BN and demonstrates the well-crystallized structure of the BNNS. Meanwhile, as shown in Fig. 3(c), the nanosheet presents a well integrity atomic structure. Furthermore, according to the geometrical relationship, the spacing between the atoms is 1.43 Å, which is quite close to the well-known length of a B–N bond of 1.44 Å. All the evidences make it clear that the sonication-assisted hydrothermal method can get high-crystallinity BNNS with a few layers (4 to 6).

Compared with the TEM images of SC-BNNS and HT-BNNS, shown in Figs. 4(a) and 4(b), the ST-BNNS appears the same high crystallinity with HT-BNNS, while the SC-BNNS is poor.

Scanning electron microscopy (SEM) analyses (Fig. 2(a)) show that the starting powders consist of flakes with the lateral size of 2–4 micrometres and thickness of hundreds of nanometres. In comparison, the sediment separated by centrifuging the solution treated with sonication-assisted hydrothermal method, contains smaller flakes, with a lateral size of < 2 μm (Fig. 2(b)). It is obvious that these flakes are consisted of many nanosheets stacked together loosely (Fig. 2(b)), compared with the pristine particles (Fig. 2(a)). Notably, the sonication-assisted hydrothermal method can overcome the intermolecular force effectively, to exfoliate the intact particle into flakes. Meanwhile, it is obvious that the sediment can be exfoliated into nanosheets furtherly to improve the yield. In addition, the method not only results in the BN particle reducting of thickness, but also the lateral size of the (002) BNNSs becomes smaller, compared with pristine particles. This is an inevitable phenomenon, as some researches have pointed out that the mechanical force can peel off BNNS from some defective areas of precursor BN particles, meanwhile reducing the lateral size in some degree [6,18]. Eventhough, the lateral size is also larger than the traditional method using high-power sonication-assisted liquid exfoliation methods.

Atomic force microscopy (AFM) was also used to study the nanosheets’ thickness and fine structure, which was unobservable in SEM and TEM clearly. Figure 5(a) shows a few flat nanosheets of ST-BNNS with thickness of about 2.4 nm (the height profile of Fig. 5(b)). A previous report shows that the AFM height of a monolayer BNNS increases from 0.33 to
1 nm owing to the trapped solvent between the BNNS and the underlying substrate [19]. Thus, the number of layers in the AFM data is less than 6, which is in good agreement with our HRTEM shown in Fig. 3(b). The AFM image also shows that the lateral size of the SH-BNNSs is largely more than 1 μm.
As shown in Fig. 6, the BNNS with the treatment of sonication exhibits the thickness of 4.8 nm, nearly 13 layers, while it treated by the hydrothermal is only 2.8 nm, about 5 layers, with the lateral size of nearly 1 μm. Interestingly, comparing Fig. 6(b) with Fig. 5(b) and Fig. 6(d), it is obvious that the rangeability of the HT-BNNS and SH-BN on the profiles is larger than the SC-BN and the distribution is nonuniformity. It is possible that the solvent permeated into the layers under the high pressure produced in the process of hydrothermal, making some bubbles on the surface, which is corresponding with our envisage.

Raman characterization is performed for the BNNS on the Si/SiO₂ substrate at the consistent condition (633 nm excitation source, 5 s for per cycle, 3 cycles). In Fig. 6, Raman spectra of SC-BNNS, HT-BNNS, and SH-BNNS show a dominant peak in the range of 1367.4–1369.6 cm⁻¹, which is assigned to the high-frequency E₂g mode (typically, single-crystal BN 1366 cm⁻¹; highly ordered pyrolytic BN 1366–1367 cm⁻¹; polycrystalline BN 1367–1374 cm⁻¹) [20]. The intensity and full width at half maximum (FWHM) of the Raman peaks can be used to evaluate the crystallinity of h-BN [20–22]. Compared with the spectrum of BNNS, treated with different methods, the peak intensities of the HT-BNNS and SH-BNNS are much stronger than it treated with the sonication, indicating that the crystalline is much higher. The FWHM of the SC-BNNS is 12.69 cm⁻¹, a little larger than the HT-BNNS (11.31 cm⁻¹) and SH-BNNS (11.62 cm⁻¹), which can be accounted for the lower crystallinity of the SC-BNNS. As the high energy of the sonication can make damage on the sheets of BNNS, at the atmosphere with high pressure and high temperature, it is not easy to destroy the lattice of the BNNS, keeping high crystallinity, which is in agreement with our TEM study, and again confirms our proposal. This suggests the exfoliating method of sonication-assisted hydrothermal is a more useful way to obtain high crystalline BNNS compared with conditional sonication treatment.

After having characterized the morphology and the crystallinity of BNNS, exfoliated by different methods, the yield was also determined and the details are as follows. The dispersions were transferred to a tube and centrifuged at 3500 rpm for 20 min. After centrifugation, the top 80 mL supernatant (out of 100 mL) was carefully collected. The dispersion was allowed to settle for one
month to subside out these undispersed particles as much as possible. The supernatant was carefully collected in a new vial with known mass and removed the solvents through freeze drying overnight. The mass of the BNNS can be determined by calculating the mass difference of the vial containing solid-state BNNS and the blank vial. Through counting, the corresponding yield is 0.88% of BNNS treated by sonication, 0.12% of HT-BNNS, and 1.68% of SH-BNNS, respectively, as shown in Fig. 8. It should be noted that a large amount of h-BN powders remained unreacted, which can be attributed to the inhomogeneous nature of solid-state reactions. The SEM images of the as-obtained products before rinsing provide direct support. These unreacted materials can be collected for subsequent exfoliation.

Based on the above analysis, it is obvious that sonication-assisted hydrothermal exfoliation is an efficient way to get high-crystallinity BNNSs with large yield. Here, we propose the key factors in the process, compared with traditional sonication and hydrothermal, respectively as follows. (1) Immerse the bulk BN into the 60% tertiary-butanol solution. As many researchers pointed out the optimal solvents are those with surface energy close to that of h-BN [23,24]. Herein, some research pointed out that the 60% tertiary-butanol solution is the best candidate to exfoliate and disperse h-BN, compared with other solution. Meanwhile, the melting boiling point is lower, and can evaporate at the bath temperature, producing high pressure under the bath temperature. (2) The tertiary-butanol and water enter into the interlayer space from the edges or defects with the assistance of sonication and the pressure. Researchers have revealed that the van der Waals force between the two water molecules is decreased with increasing temperature in a condition of high temperature and high pressure. This indicates the water molecules are force-free and have vigorous Brownian motion of molecules under sonication-assisted hydrothermal conditions [25–28]. Interestingly, H₂O is a polar molecule of a V-shaped structure and the surface tensions of water are close to h-BN in hydrothermal conditions, suggesting that water molecules can be adsorbed on BNNSs, and separate the flakes. What is more, at the bath temperature of 355 K, the interaction energy is calculated to be −8.6 kcal/mol, compared with −10.6 kcal/mol at 297 K [29]. The sonication furtherly weaks the interlayer bonding and reinforces the vigorous Brownian motion of molecules, improving more solution into the layers under the high pressure. (3) The cavitation effect of sonication acted on the H₂O facilitates exfoliation of h-BN sheets from the surface of h-BN powders [30,31]. During the sonication, shear forces and cavitation, i.e., the growth and collapse of the micrometer-sized bubbles or void in the liquids due to pressure fluctuations, act on the inter-sheet attractive forces. Exfoliated BNs are dispersed by the 60% tertiary-butanol solution, avoiding the stack of layers.

In summary, we have demonstrated a novel method to produce BNNS via sonication-assisted hydrothermal treatment, different from traditional sonication or hydrothermal method. Through combinating of the hydrothermal treatment and the sonication by one step, the high-quality BNNSs are prepared with a high yield. The exfoliation mechanism is put forward and reveals that the exfoliating process can be applicable to other two-dimensional materials.
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