Pressure-induced high-temperature superconductivity retained without pressure in FeSe single crystals

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Contributed by Ching-Wu Chu, June 10, 2021 (sent for review May 13, 2021; reviewed by Qiang Li and James S. Schilling)

To raise the superconducting-transition temperature (Tc) has been the driving force for the long-sustained effort in superconductivity research. Recent progress in hydrides with Tc up to 287 K under pressure of 267 GPa has heralded a new era of room temperature superconductivity (RTS) with immense technological promise. Indeed, RTS will lift the temperature barrier for the ubiquitous application of superconductivity. Unfortunately, formidable pressure is required to attain such high Tc. The most effective relief to this impasse is to remove the pressure needed while retaining the pressure-induced Tc without pressure. Here, we show such a possibility in the pure and doped high-temperature superconductor (HTS) FeSe by retaining, at ambient pressure via pressure quenching (PQ), its Tc up to 37 K (quadrupling that of a pristine FeSe at ambient) and other pressure-induced phases. We have also observed that some phases remain stable without pressure at up to 300 K and for at least 7 d. The observations are in qualitative agreement with our ab initio simulations using the solid-state nudged elastic band (SSNEB) method. We strongly believe that the PQ technique developed here can be adapted to the RTS hydrides and other materials of value with minimal effort.

FeSe | high-temperature superconductivity | high pressure | pressure quench | retention

The vast impact of room temperature superconductivity (RTS) on humanity is limited only by the imagination. Recent reports show that RTS is indeed within reach, although only under high pressure. For instance, superconducting-transition temperatures (Tcs) above 200 K have been reported in unstable molecular solids (hydrides), i.e., up to 203 K in H2S under 155 GPa (1, 2), up to 260 K in LaH3 under 190 GPa (3–5), up to 287 K in C–H–S under 267 GPa (6), and potentially well above room temperature in La–H under 158 GPa after thermal cycling (7); earlier, Tc up to 164 K was reported in the stable cuprate high-temperature superconductor (HTS) Bi2Sr2CaCu2O8+δ under 31 GPa (8, 9). While record-high Tcs reported to date fall into practical cryogenic regimes for applications, the HP required to attain these superconducting states renders them impractical for significant applications or for scientific inquiries. The challenge is not restricted to further increasing the superconducting transition temperature under extreme conditions and must now include concentrated efforts to lower, and better yet remove, the applied pressure required. This work addresses directly such a challenge by demonstrating our successful retention of pressure-enhanced and/or -induced superconducting phases in pressure quenched (PQed) HTSs via PQ (Fig. 1B) at a chosen quench pressure (Pq) and quasi thermal temperature (Tq) below which the pressure from which the pressure is rapidly removed to ambient. Tc is the temperature at which the pressure is removed, and it remains unchanged during the PQ process. We have successfully retained a pressure-enhanced SC phase with a Tc up to 37 K at Tq = 4.15 GPa and Tp = 4.2 K in the FeSe and a pressure-induced SC phase with a Tc up to 26.5 K at Pq = 6.32 GPa and Tp = 4.2 K in the non-SC Cu-doped FeSe. We have also retained the insulating phase induced by pressure above ~9 GPa in both samples via PQ. The pressure-quenched (PQed) high-Tc phases have also been found to be stable up to ~200 K and of such energy barriers, lattice and/or electronic, one may therefore be able to stabilize the metastable phase or the “supercooled” state at atmospheric pressure via rapid pressure quenching (PQ) and/or temperature quenching. The energy barrier may be fortified by chemical doping; ionic liquid gating (14); a proper thermodynamic path; and introduction of strains (15), defects (16), or pressure inhomogeneity (17). The pressure-enhanced or -induced SC phase with a high Tc may be considered metastable or supercooled and may be stabilized. Here, we report the successful retention of pressure-enhanced and -induced SC phases at ambient pressure in the Fe-based HTSs via PQ (Fig. 1B) at a chosen quench pressure (Pq) and quasi thermal temperature (Tq) below which the pressure is rapidly removed to ambient. Tc is the temperature at which the pressure is removed, and it remains unchanged during the PQ process. We have successfully retained a pressure-enhanced SC phase with a Tc up to 37 K at Tq = 4.15 GPa and Tp = 4.2 K in the FeSe and a pressure-induced SC phase with a Tc up to 26.5 K at Pq = 6.32 GPa and Tp = 4.2 K in the non-SC Cu-doped FeSe. We have also retained the insulating phase induced by pressure above ~9 GPa in both samples via PQ. The pressure-quenched (PQed) high-Tc phases have also been found to be stable up to ~200 K and of such energy barriers, lattice and/or electronic, one may therefore be able to stabilize the metastable phase or the “supercooled” state at atmospheric pressure via rapid pressure quenching (PQ) and/or temperature quenching. The energy barrier may be fortified by chemical doping; ionic liquid gating (14); a proper thermodynamic path; and introduction of strains (15), defects (16), or pressure inhomogeneity (17). The pressure-enhanced or -induced SC phase with a high Tc may be considered metastable or supercooled and may be stabilized. Here, we report the successful retention of pressure-enhanced and -induced SC phases at ambient pressure in the Fe-based HTSs via PQ (Fig. 1B) at a chosen quench pressure (Pq) and quasi thermal temperature (Tq) below which the pressure is rapidly removed to ambient. Tc is the temperature at which the pressure is removed, and it remains unchanged during the PQ process. We have successfully retained a pressure-enhanced SC phase with a Tc up to 37 K at Tq = 4.15 GPa and Tp = 4.2 K in the FeSe and a pressure-induced SC phase with a Tc up to 26.5 K at Pq = 6.32 GPa and Tp = 4.2 K in the non-SC Cu-doped FeSe. We have also retained the insulating phase induced by pressure above ~9 GPa in both samples via PQ. The pressure-quenched (PQed) high-Tc phases have also been found to be stable up to ~200 K and...

**Significance**

As room temperature superconductivity (RTS) has been reported recently in hydrides at megabar pressures, the grand challenge in superconductivity research and development is no longer restricted to further increasing the superconducting transition temperature under extreme conditions and must now include concentrated efforts to lower, and better yet remove, the applied pressure required. This work addresses directly such a challenge by demonstrating our successful retention of pressure-enhanced and/or -induced superconducting phases and/or semiconducting phases without pressure in single crystals of superconducting FeSe and non-superconducting Cu-doped FeSe. The pressure-quenching technique developed in this work offers the possibility of future practical application and the unraveling of RTS recently detected in hydrides but only under high pressures.

Author contributions: L.D., Z.W., and C.-W.C. designed research; L.D., T.B., R.D., M.G., and D.R. performed research; Y.X., B.G., X.L., K.Y., Z.W., Y.M., and P.D. contributed new reagents/analytic tools; L.D., T.B., and C.-W.C. analyzed data; L.D. and C.-W.C. wrote the paper; and L.D. and C.-W.C. directed the project.

Reviewers: Q.L., Brookhaven National Laboratory; and J.S.S., Washington University in St. Louis.

The authors declare no competing interest.

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This article contains supporting information online at https://www.pnas.org/lookup/suppl/doi:10.1073/pnas.2108938118/-/DCSupplemental.

Published July 7, 2021.
In the present study, we have chosen single crystals of the SC FeSe (18) and the non-SC Cu-doped FeSe (19) as model HTSs due to their simple structure and chemistry, as well as their large Tc variation under pressure (20, 21) and their important role in unraveling HTSs (22–24). Furthermore, the iron-chalcogenide superconductors have attracted broad interest for applications from high-field magnets to quantum information science. For example, the Majorana zero modes reported in iron-chalcogenide superconductors can potentially be used for building topological qubits (25). The normalized resistance of FeSe and Cu-doped FeSe at 300 K as a function of pressure [R(P)]/R(0)] during pressure increasing and decreasing is displayed in SI Appendix, Fig. S1, which shows a clear hysteresis, suggesting that PQ may be possible since thermal hysteresis may provide the energy barrier (Fig. L4) to retain the HP-induced phases. Preliminary boundaries of the orthorhombic (O)–tetragonal (T)–hexagonal (H) phase transitions of FeSe previously reported (20, 26) are also shown for later discussion. The T–O transition is suppressed from ~90 K at ambient pressure to below 4.2 K at ~2 GPa, as indicated by the dashed line at Left in the same figure. At ambient pressure, R(T) of FeSe shows a sharp SC transition at 9.3 K (SI Appendix, Fig. S2). The transition broadens under pressure, so the Tc(P) cited hereafter refers to the onset temperature as defined in SI Appendix, Fig. S2. Fig. 2 (blue squares) displays the Tc variation of FeSe with PA: It increases slowly from ~9 K at ambient pressure to ~15 K below 1.9 GPa; suddenly jumps to ~32 K at 1.9 GPa, coinciding with the O–T transition; continues to rise with a broad peak at ~40 K around 4 GPa; but finally becomes insulating above ~8 GPa as the H phase sets in.

To retain at ambient pressure the above pressure-enhanced Tc of FeSe, we have developed a technique to PQ the sample at different PQs and TQs by rapidly removing the PA, under which a desired Tc has been first attained, from the sample in the diamond anvil cell, as shown in Fig. 3 A–F. The temperature-dependent resistance of FeSe at different PQs normalized to those at 70 K, R(T,PA)/R(70 K,PA), near the superconducting transitions are exemplified in Fig. 3 A and B for PQA = 4.15 GPa (close to maximum Tc ~40 K in the tetragonal phase), and 11.27 GPa (non-SC in the hexagonal phase), respectively. By following different thermal and pressure protocols as specified in the captions, they demonstrate the generation or destruction of the HP SC phase at PA (blue), the retention at ambient pressure of the PQsed (at 4.2 K) HP SC phase (red), and the thermal annealing effect up to 300 K on the PQsed (at 4.2 K) HP phase to ascertain its retention (orange), all carried out sequentially.

As is evident from Fig. 3A, the Tc of the FeSe sample has been enhanced from ~9 K at ambient pressure to ~39 K under 4.15 GPa (blue). After PQ at 4.15 GPa and 4.2 K, a SC transition with a Tc ~37 K is detected at ambient pressure (red). To show that the 37 K–Tc is indeed attained by PQ, we heated the sample up to 300 K before cooling it back down to 4.2 K and found that the PQsed SC transition at 37 K is annealed away and replaced by its pre-PQsed one, although at a higher Tc ~20 K (orange) rather than ~9 K, presumably because of an unknown irreversible residual strain effect in the sample (27). Fig. 3B shows that FeSe at 11.27 GPa displays a non-SC transition as expected (blue), as does the PQsed sample (red). However, the sample regains its SC transition with a Tc ~20 K after the PQsed phase is annealed off after being heated up to 300 K (orange). To demonstrate the metastability of the PQsed SC phases, the SC transition PQsed at PQA = 4.13 GPa and TQ = 4.2 K upon sequential thermal cycling to higher temperatures is shown in Fig. 3C. The transition smoothly shifts downward and becomes sharper due to possible reduced fluctuations at lower temperature and/or the possible improved strain condition of the sample upon thermal annealing at higher temperatures. The sudden downward shift in the overall SC transition by ~10 K after heating up to ~200 K implies that the PQsed phase transforms to the pre-PQsed FeSe phase (with strain) and is stable up to 200 K. All Tcs of the PQsed phases examined at different PQs and TQ = 4.2 K are summarized in Fig. 2 (red circles).
As mentioned earlier, the PQed phase is metastable, and thus should depend on $P_A$ and $T_Q$ and detailed electronic and phonon energy spectra of the materials examined. We have therefore repeated the PQ experiments on FeSe by raising only the $T_Q$ to 77 K (Fig. 3D–F). Fig. 3D shows that the $T_c$ of FeSe before PQ has been enhanced to ~37 K at 5.22 GPa (blue); upon PQ, a $T_c$ ~24 K is retained at ambient pressure (green) in contrast to the 37 K when $T_Q = 4.2$ K, as shown in Fig. 3A; and the transition returns to ~14 K on cooling after warming to 300 K, showing that the 24 K transition is associated with the PQed phase. Fig. 3E shows that FeSe becomes insulating at 11.12 GPa (blue); the phase is retained at ambient pressure by PQ (green); and the PQed phase remains after heating to 300 K, suggesting that this PQed non-SC phase is stable up to 300 K. The effect of systematic thermal cycling with increasing temperatures on the PQed phase at $P_Q = 5.22$ GPa and $T_Q = 77$ K is shown in Fig. 3F. All $T_s$ of the PQ phases examined at different $P_Q$s and $T_Q$s = 77 K are also summarized in Fig. 2 (green diamonds). They are all lower than those quenched at various $P_Q$s and $T_Q$s in general agreement with the competition between the instability of the SC state and thermal excitation.

To demonstrate that the retained SC state after PQ in pure FeSe at ambient pressure is not associated with the superconductivity of the pristine FeSe at ambient pressure, we have repeated the PQ experiment on two non-SC Cu-doped FeSe samples (Fe$_{1.01}$, CuSe with $x = 0.03$ and 0.035; the $x = 0.03$ sample is discussed below unless otherwise noted). As shown in SI Appendix, Fig. S2B, Cu-doped FeSe is not SC above 1.2 K below 1.2 GPa (19, 21). Under pressure (Fig. 4, blue squares), it abruptly becomes SC with a $T_c$ ~20 K at 3.11 GPa (Inset, SI Appendix, Fig. S2B); $T_c$ continues to increase with increasing $P_A$ and peaks at ~27 K under 6.23 GPa; and at 9.65 GPa, only trace superconductivity was detected down to 1.2 K. Following the same protocols as those for the pure FeSe, we performed PQ on Cu-doped FeSe at different $P_Q$s and $T_Q$s, as exemplified by Fig. 5A–F. Two examples of $R(T,P_Q)/R(50$ K, $P_A)$ for Cu-doped FeSe are given in Fig. 5A for $P_Q$s = 6.32 GPa and 6.16 GPa (close to maximum $T_c = 27$ K) PQed at $T_Q = 4.2$ K and 77 K, respectively; and in Fig. 5B for 9.65 GPa (non-SC) PQed at $T_Q = 77$ K. As is evident from the $R(T,P_Q)/R(50$ K, $P_A)$ in Fig. 5A, $P_A = 6.45$ GPa has induced a SC state in the non-SC Cu-doped FeSe with a $T_c$ ~26 K (navy and blue); this SC state has been PQed at $P_Q = 6.16$ GPa and $T_Q = 4.2$ K (red) and at $P_Q = 6.32$ GPa and $T_Q = 77$ K (green), respectively. Disappearance of the SC phase after thermal cycling up to 300 K (Fig. 5A, orange and brown) demonstrates that the SC states induced by $P_A = 6.45$ GPa have been retained at ambient pressure with $T_c = 26$ K via PQ at 4.2 and 77 K, respectively. As shown in Fig. 5B, $P_A = 9.65$ GPa turns the sample to an insulating state (blue); upon PQ at $T_Q = 77$ K, it remains insulating (green); and the sample stays in the non-SC state after thermal cycling to 300 K (orange), suggesting that the insulating state PQed at $T_Q = 77$ K is stabilizable up to 300 K. The thermal stability ranges of the PQed SC states at $P_Q = 6.08$ GPa and $T_Q = 4.2$ K and at $P_Q = 5.95$ GPa and $T_Q = 77$ K are shown in Fig. 5C and D, respectively. They show that the state PQed at a lower $T_Q$ possesses a wider thermal stability range. The anomalies observed in $R(T)$ upon warming right after PQ (Fig. 5E) correlate qualitatively with the thermal stability of the PQed phases (Fig. 5C and D). Fig. 5F demonstrates that the PQed SC phase at $P_Q = 6.67$ GPa and $T_Q = 77$ K remains unchanged for at least 7 d after thermal cycling between 50 and 4.2 K. All PQed $T_s$ of Cu-doped FeSe are summarized in Fig. 4. Unlike in their pristine unpressurized state, the two different Cu-doped FeSe samples both behave similarly to FeSe under pressures, but with
their phase boundaries shifted to higher values, as displayed in Figs. 2 and 4 and SI Appendix, Fig. S3, due to the Cu-doping effect. While PQ works for both pure and Cu-doped FeSe in retaining superconductivity between 77 and 220 K as shown; (E) R(T) at ambient pressure for the sample PQed at 6.32 GPa and T\textsubscript{Q} = 77 K (green) and at T\textsubscript{Q} = 120 K (purple); and (F) repeated thermal cycling at ambient pressure from 50 K for the sample PQed at 6.67 GPa and 77 K.

Fig. 4. \(T_c\) as a function of \(P_A\) or \(P_Q\) for single-crystalline Cu-doped FeSe. \(T_c\) at \(P_A\) and at ambient pressure for the samples PQed at \(T_Q\) and \(T_Q = 4.2\) K (red circles) and at \(T_Q = 77\) K (green diamonds), respectively.

Fig. 5. Pressure quenching (PQ) the single-crystalline Cu-doped FeSe. R(T)/R(50 K) under \(P_A\) and at ambient pressure after PQ, and testing the stability of the PQed phases: (A) at \(P_A\) = 6.16 GPa (navy) and 6.32 GPa (blue), and at ambient pressure after PQ at 6.16 GPa and 77 K (green) and at 6.32 GPa and 4.2 K (red), and on cooling after warming to 300 K (orange and brown); (B) at \(P_A\) = 9.65 GPa (blue), at ambient pressure after PQ at 9.65 GPa and 77 K (green), and on cooling after warming to 300 K (orange); (C) at ambient pressure after PQ at 6.08 GPa and 4.2 K, warmed to 25 K and sequentially cooled from different temperatures between 25 and 220 K as shown; (D) at ambient pressure after PQ at 5.95 GPa and 77 K sequentially cooled from different temperatures between 77 and 220 K as shown; (E) R(T) at ambient pressure for the same sample subjected to different PQ conditions: \(P_Q\) = 6.31 GPa and 6.32 GPa (blue), at ambient pressure after PQ at 5.95 GPa and 77 K (green), and on cooling after warming to 300 K (orange); and (F) repeated thermal cycling at ambient pressure from 50 K for the sample PQed at 6.67 GPa and 77 K.
FeSe returning to the orthorhombic phase from the hexagonal phase at ambient pressure and 300 K, which is consistent with our experiments at $P_O = 11.12$ GPa and $T_Q = 77$ K shown in Fig. 3E.

**Conclusions**

We have demonstrated that the pressure-enhanced or -induced superconducting phases with high $T_c$ and the pressure-induced semiconducting phases in FeSe and Cu-doped FeSe can be stabilized without pressure by PQ at chosen pressures and temperatures. These PQed phases have been shown to be stable at up to 300 K and for up to at least 7 d depending on the quenching conditions. The observations raise the hope that the recently reported RTS in hydrides close to 300 GPa may be retained without pressure, making possible the ubiquitous applications of RTS envisioned.

**Materials and Methods**

**Sample Preparation.** Single crystals of $\text{Fe}_{1.01-x}\text{Cu}_x\text{Se}$ ($x = 0, 0.03$, and 0.035) were grown using the chemical vapor transport method (34). Stoichiometric Fe (99.9%; Alfa Aesar), Cu (99.9%; Alfa Aesar), and Se (99.5%; Alfa Aesar) powders were thoroughly mixed and loaded into a quartz tube. AlCl$_3$ (99%; Alfa Aesar), Cu (99.9%; Alfa Aesar), and Se (99.5%; Alfa Aesar) powders grown using the chemical vapor transport method (34). Stoichiometric Fe (99.9%; Alfa Aesar), Cu (99.9%; Alfa Aesar), and Se (99.5%; Alfa Aesar) powders were added as the transport agents. After the evacuated quartz tube was sealed, it was placed into a two-zone tube furnace, in which the temperatures of the hot and cold positions were maintained at 420 and 330 °C, respectively. After 20 d, single crystals with an average size of $3 \times 3 \times 0.1$ mm$^3$ were grown around the region of the quartz tube's cold zone.

**Electrical Transport Measurements under Pressure.** For resistivity measurements conducted in this investigation, pressure was applied to the samples using a Mao-type symmetric diamond anvil cell (35) with a cuvet size of 500 μm. The gaskets are made from T301 half-hard stainless-steel sheets with thickness of 300 μm. Each gasket was pretrained to ~20–40 μm in thickness and was insulated with Stycast 2850FT. The sample's chamber diameter is ~230–250 μm, where either sodium chloride or cubic boron nitride is used as the pressure-transmitting medium. Samples were cleaved and cut into thin squares with a diagonal of ~200 μm and thickness of ~20 μm. The pressure was determined using the ruby fluorescence scale (36) or the diamond Raman scale (37) at room temperature. The samples' contacts were arranged in a Van der Pauw configuration, and data were collected using a Keithley 6221/2182A Delta Mode System. Measurements were conducted in a homemade cooling system that can be cooled to 1.2 K by pumping on the liquid-helium space. PQ was performed by releasing the screws at target temperatures down to 4.2 K with a small residual pressure $P_R < 0.2$ GPa to maintain the electrical connectivity for resistivity measurements, and the $P_R$ was measured at room temperature.

**Theoretical Calculations.** Our calculations were performed within the framework of density functional theory via the generalized gradient approximation GGA + U method implemented in the Vienna ab initio simulation package (VASP) (38). The electron-ion interactions were represented by means of the all-electron projector augmented wave method (39), where 3$d^4s^2$ and 4$d^5s^2$ are treated as the valence electrons for Fe and Se, respectively. We used the Dudarev implementation (40) with on-site coulomb interaction $U = 5.0$ eV and on-site exchange interaction $J = 0.8$ eV (41) to treat the localized 3d electron states. The Perdew–Burke–Ernzerhof functional in the generalized gradient approximation (GGA) was used to describe the exchange-correlation potential (42, 43). The plane-wave energy cutoff of 400 eV and a dense $k$-point grid of spacing $2\pi \times 0.03$ Å$^{-1}$ in the Monkhorst–Pack scheme were used to sample the Brillouin zone. Structural relaxations were performed with forces converged to less than 0.05 eV Å$^{-1}$. To determine the energy barriers, we used the SSNEB (28) implemented in VASP. The NEB path was first constructed by linear interpolation of the atomic coordinates and then relaxed until the forces on all atoms were <0.05 eV Å$^{-1}$. Seven images were simulated between the initial and final states.

**Data Availability.** All study data are included in the article and/or supporting information.

**ACKNOWLEDGMENTS.** We thank Prof. L. L. Sun, C. Huang, and J. Guo at the Institute of Physics, Chinese Academy of Sciences, for discussions. The work performed at the Texas Center of Superconductivity at the University of Houston is supported by US Air Force Office of Scientific Research Grants FA9550-15-1-0236 and FA9550-20-1-0068, the T. L. L. Temple Foundation, the John J. and Rebecca Moores Endowment, and the State of Texas through the Texas Center for Superconductivity at the University of Houston. The FeSe and Cu-doped FeSe single-crystal growth work at Rice University is supported by the US Department of Energy, Basic Energy Sciences, under Contract DE-SC0012311 (P.D.).

![Fig. 6](https://doi.org/10.1073/pnas.2108938118)
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