Role of starch in the ceramic powder synthesis: a review

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Received: 4 June 2020 / Accepted: 24 August 2020 / Published online: 10 September 2020 © The Author(s) 2020

Abstract
The main goal of this work was to demonstrate a broad range of starch applications in ceramic powder synthesis, and to point out that starch and its derivatives can successfully replace polyvinyl derivatives commonly used in ceramic formulations, as they are water-soluble saccharides, burn out more easily, reduce the exothermicity of the combustion reaction, and allow to receive fine and uniform ceramic powders. Starch is an eco-friendly and easily transformable natural carbohydrate polymer that is found in a variety of applications in ceramics and materials science. In this paper, we review the influence of starch on the physicochemical properties of the ceramic powders obtained by means of various synthesis methods (sol–gel method, combustion process, combined methods, and hydrothermal synthesis). Starch plays a differentiated role in each of these methods, i.e., it is a polymerizing agent in sol–gel method, a fuel in combustion process, and a substrate for carbon spheres in hydrothermal synthesis.

Graphical Abstract

Keywords Starch · Sol–gel synthesis · Combustion process · Sol–gel combustion process · Fuel/oxidation ratio

Highlights
- Starch can replace polyvinyl derivatives used in ceramic formulations.
- Starch helps to control particle size of the powders and prevents agglomeration.
- Starch helps to create a more compact and porous structure in the sol–gel method.
- Starch is a highly energetic fuel that provides energy during combustion.

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1 Introduction

The properties of the ceramic products, therein ceramic powders, are largely affected by the phase and morphological characteristics of the initial raw material powders, such as the level of aggregation, purity, particle shape, and uniformity in chemical and phase composition. Another factor is the method of the powder preparation/synthesis, which is divided into three categories, i.e., solid-phase reactant, liquid-phase reactant, and gas-phase reactant. These groups of methods are used mainly for the preparation of ceramic powders that can be subjected to sintering. The last one factor concerns the physicochemical properties of the supporting substances and their role in the preparation/synthesis process. In this case, we can mention carbon (e.g., graphite or carbon black), organic polymers (e.g., polyethylene or polyacrylic acids), or natural biopolymers (e.g., wood flour or starch). Among them, the broadest range of applications shows starch [1, 2].

Instead of using nonrenewable, environmentally hazardous, and costly petrochemical-derived additives, which are difficult to treat from the point of view of recycling, starch and its derivatives can offer the ceramic industry a set of applications, multifunctionality, differentiated properties, aqueous processing, and easy burnout. It is worth to point out that starch belongs to the so-called renewable biological resources. Its scope of production fluctuates under the influence of various factors (economy and climate), yet it can be raised considerably in a relatively short period of time [3]. What is more important, products of starch (polysaccharides) pyrolysis are comparable to those of polyvinyl derivatives, commonly used in ceramic formulations, and in comparison to them, water-soluble saccharides burn out more easily and cleaner. Starch can act as the polymerizing, chelating, or gelling agent in sol–gel methods. It can be a substrate for carbon spheres in hydrothermal synthesis, as well as pore-forming agent in the manufacturing of porous ceramic and bioceramic powders. Apart from that, it is an excellent source of carbon and can provide energy in the combustion synthesis process. It should also be mentioned that starch and its derivatives can be applied as binders, rheology modifiers, or even strengthening agents in traditional ceramics manufacturing [4–7].

From the chemical point of view, starch is a carbohydrate polymer consisting of glucose units connected by α-glycosidic bonds. It constitutes the main food-reserve material of a majority of higher plants. The most abundant plants capable of synthesizing starch are cereals (corn, wheat, rice, barley, oats, rye, and sorghum), tubers (potato, tapioca, and arrowroot), palm trees (e.g., sago), as well as leguminous plants (e.g., peas) [8]. It occurs in the form of small granules of 2–150 µm that are insoluble in water (because of the highly ordered and densely packed glucan chains that create its microstructure). It is also worth to mention that plant origins strongly influence the physicochemical properties of starch, such as structure, shape, size, gelling, and swelling ability (Fig. 1) [9, 10].

Starch processing begins with mechanical removing of its granules from the plant material. It involves processes like grinding, sieving, washing, and filtration. The next step is the physical, chemical, or biological treatment that causes starch transformation into modified derivatives with new properties like, e.g., water solubility. For example, acid hydrolysis breaks long polymer starch chains into shorter ones, as follows: amylopectin, erythrodextrin, acroldextrin, maltose, and glucose. In turn, heating in the acid environment results in dextrans or so-called British gums that are water soluble [11–17].

This review aims at realization of the great importance of starch in the preparation of ceramic materials, therein ceramic powders. In our previous work [18], we have described use of starch and its derivatives in different sections of ceramics, starting with traditional ceramics and ending with bioceramics. From that time, because inter alia of its eco-friendly properties, starch found a lot more interest in described areas, especially ceramic powder preparation. For this reason, because some older patents and articles could lose their topicality, it seemed reasonable to resume this topic again, and this review is based only on articles about ceramic powder synthesis written after 2005. We have focused on the role and importance of starch in

Fig. 1 SEM images of a tapioca, b potato, and c corn starch granules (magnification ×1000) [18]
terms of its applications in methods leading to the production of ceramic powders. Starch is an eco-friendly and easily transformable carbohydrate polymer that plays a differentiated role in the ceramic powder synthesis.

2 Role of starch in different methods of ceramic powder synthesis

2.1 Sol–gel method

Sol–gel process is a method of the production of inter alia ceramic powders. It consists in the slow dehydration of the hydroxide sol of a given powder, resulting in a gel, which then undergoes the calcination process and turns into fine and uniform powder. The advantage of this method is the relatively low production cost and the possibility of obtaining various types of materials. On the other hand, the disadvantages include the difficulties associated with the controlling of the particle size and the degree of its agglomeration. This problem can be minimized by applying either various capping agents that block the growth of grains or polymerizing agents that support gelling process of the system. Currently, the most commonly used for this purpose are polymers of natural origin, mainly starch and its derivatives (Fig. 2) [19–30].

Zak et al. [19] used the sol–gel method with the presence of starch as the stabilizer to control the cation mobility and growth of the resulting ZnO nanoparticles. According to the increasing calcination temperatures (400–600 °C), ZnO nanopowders of 28–51 nm and of hexagonal structure were obtained. Balcha et al. [20] obtained similar results. Using a calcination temperature of 600 °C, they got crystallites with a size of 28 nm and of hexagonal wurtzite structure. They also proved that the sol–gel method is better than precipitation, because the obtained crystallites are smaller and their photocatalytic properties are better. In turn, de Almeida et al. [21] received ZnO crystallites of 20 nm, using the calcination temperature of 500 °C by 2 h. They also concluded that the cassava starch used as the stabilizing agent inhibits the crystal growth, thanks to the interaction of the C–O groups of the glucose (resulted from the starch hydrolysis) and zinc cations.

Darroudi et al. [22] produced crystalline nanoceria powders by the sol–gel method in an aqueous solution. Long-chained starch terminated the growth of nanoceria particles during the reaction and stabilized them by covering the powder nanoparticles with hydroxyl groups as a capping agent. The resulting cerium hydroxide samples were further calcined in the temperature range 200–600 °C for 2 h to obtain CeO₂ nanopowders of about 6 nm with a cubic

![Scheme of the starch-doped sol–gel preparation of the ceramic nanopowders](image-url)
fluoride structure. Ferreira et al. [23] obtained nanoceria powders of ~8–16 nm and of fluoride structure using cassava starch as the chelating agent and calcination temperature 200–500 °C for 1 h.

Alishah et al. [24] investigated the antimicrobial properties of the copper oxide nanopowders produced via the starch-assisted sol–gel method. The temperature of the gel formation in this process was 65 °C, whereas calcination temperature was 400 °C. The utilized carbohydrate helped to received stable and spherical-shaped nanopowders, whose average size was 54 nm.

Khodadadin [25] compared the structural, optical, and photocatalytic properties of the undoped and Nd–Ce-codoped titanium dioxide, where starch was used as the “green” modifier that prevents agglomeration, limits excessive grain growth, and in conclusion modifies the photocatalytic activity of the synthesized nanopowders. He also revealed that small amounts of dopants (up to 1 mol%) improve the photocatalytic activity because they reduce the bandgap. Higher dopant amount (2 mol%) means occupation of the active surface sites by the Ce and Nd particles, and as a result, disturbs the photocatalytic effect.

Zhang et al. [26] compared the capping/stabilizing properties of acrylamide and starch/glutaric dialdehyde in sol–gel synthesis of alumina nanopowders. Both of the mentioned systems helped to form a matrix (without any initiators and cross-linking agents, just self-polymerization in temperature of about 100 °C) for the entrapped metal ions and gave rise to alumina particles during heating. In both cases, the formed gel was dried for 36 h at a temperature of about 95 °C and then calcined at various temperatures from 350 to 1200 °C. The difference was in the heating temperatures in which alumina phases started to occur. For acrylamide, a g-Al2O3 phase began to appear already at a temperature of 650 °C, whereas for the starch/glutaric dialdehyde it was 804 °C. The phase system from g to a started at 1100 °C for both systems, and a pure a-Al2O3 phase appeared at 1200 °C. What speaks more for the starch/glutaric aldehyde system is that in this configuration, this monomer is less toxic than acrylamide itself.

Koferstein et al. [27] synthesized BaCeO3 nanopowder in the sol–gel reaction with the use of starch as a polymerizing agent. The obtained (BaCe) gel, after heating up to 920 °C, transformed to pure BaCeO3, which was characterized by a specific surface area of 15.4 m²/g and a crystallite size of about 41 nm. This method was also used to obtain LaFeO3 nanopowders exhibiting improved magnetic and optical properties [28, 29]. Calcination of (LaFe) gel at 570 °C allowed to obtain the pure phase of LaFeO3 powder with a specific surface area of 25.7 m²/g and a crystal size of 37 nm, while increasing the temperature to 1000 °C, Tizro and Kamali [30] used the sol–gel method to obtain PbTiO3 nanoparticles subsidized with silver. In addition to starch, glucose and lactose were used as the gelling agents. Analyzing scanning electron microscope (SEM) images, it was found that in the case of glucose, PbTiO3 powder was compacted into micrometric-size agglomerates. Using lactose, the number of agglomerates was reduced, and nanoparticles in the range of 200–400 nm were obtained, whereas in the case of starch, spherical particles with grain sizes of 30–50 nm were obtained.

Starch, thanks to its structure, i.e., long polymer chains, single-helix structure, and a-glucose units, covers the surface of nanograins by hydroxyl groups, becoming the capping agent and matrix for the nascent powders, preventing their excessive growth. In addition, thanks to the presence of a large number of hydroxyl groups that have the same properties as those coming from alcohols, starch can be used as the oxidation agent for metal salts, i.e., it is a good polymerization agent (Table 1).

2.2 Combustion process

Combustion synthesis, i.e., self-propagating high-temperature synthesis, is one of the most efficient and economic way of obtaining ceramic powders, due to its simple

| Synthesized powder | Role of starch | Temperature of calcination [°C] | Crystallite size [nm] | References |
|--------------------|---------------|-------------------------------|----------------------|------------|
| ZnO                | Stabilizer    | 400–600                       | 20–51                | [19], [20], [21] |
| CeO2               | Capping (chelating) agent/stabilizer | 200–600                  | 6–16                  | [22], [23] |
| CuO                | Stabilizer    | 400                           | 54                   | [24]       |
| TiO2               | Stabilizer    | 500                           | 28–40                | [25]       |
| Al2O3              | Capping agent/stabilizer | 350–1200                  | 3–20                  | [26]       |
| BaCeO3             | Polymerizing agent | 920                        | 41                   | [27]       |
| LaFeO3             | Polymerizing/complexing agent | 570–1000                | 37–130                | [28], [29] |
| PbTiO3             | Capping agent | 700                           | 30–50                | [30]       |
introduces high reaction temperature and specific heating/cooling rate, which helps to control the microstructure of the powders. In typical combustion synthesis, the first step involves the reaction between an oxidant precursor (usually metal nitrate) and a fuel (organic derivative) that is a reagent reducer. Energy produced as a result of the reaction sustains a further process, resulting in the ignition of reagents and the formation of a ceramic powder. The resultant products are metal oxide or its spinel, carbon dioxide, water vapor, and nitrogen. The crucial parameters here are the type of the combustion fuel and fuel-to-oxidation ratio. Selecting the fuel, we should remember about its enthalpy of combustion. The lower this parameter is, the lower the exothermicity of the combustion process, and the lower the temperature at which the selected powder is synthesized. In the past, hydrazide-type compounds have been used, but nowadays, because of its nontoxicity, mainly urea, carbon, and polysaccharides are used, among which the most popular is starch. For example, combustion enthalpy of urea (DHf) is −147.26 kcal/mol, whereas that for starch is −401.57 kcal/mol. Considering the ratio of fuel to the oxidant, three cases are possible. Stoichiometric (described above), fuel lean, in which oxygen is in excess and it appears in exhaust gas, and fuel rich, in which oxygen is deficient, so a portion of atmospheric oxygen is necessary to complete the reaction [31–33].

Tahmasebi and Paydar [32] studied the effect of starch addition on the solution combustion synthesis of Al2O3–ZrO2 powder. Because of the large heat of formation of starch molecules, they replaced urea with starch in the mixture of fuels, and in this way, reduced the exothermicity of the combustion reaction. This prevented the sintering of particles and resulted in fine agglomerates of the Al2O3–ZrO2 composite powder crystallites with a diameter less than 20 nm.

Qui et al. [34] examined the possibility of the production of AlN powder from combustion synthesis precursors, such as carbon black, glucose, sucrose, citric acid, and water-soluble starch. When the carbon source was carbon black, pure AlN powder was found in the sample calcined at 1500 °C, whereas for the precursors based on water-soluble organics, pure AlN powder was already found in the samples calcined at 1400 °C. Comparing the particle size and shape, it was found that AlN powders synthesized with the use of inter alia starch, had fine spherical shape and particle size of about 100 nm, whereas those based on carbon black were larger and their size distribution was wider.

Bai et al. [35] used the solution combustion synthesis to obtain MgO powders. The fuel-to-oxidation ratio was the crucial factor and because of that, various redox mixtures with starch-to-magnesium-nitrate mole ratio of 5.5:12, 5:12, 4.5:12, and 4:12 were examined. During the combustion process, nanoporous agglomerates were formed and the average crystalline size was about 8 nm. This value was growing up together with the starch content. They also used [33] the same method to obtain MgAl2O4 powders. As the fuel for the process, they were testing urea, glycine, and starch in various combinations and ratios. Their tests confirmed the positive effect of starch addition (0–35.6 wt%) on the specific surface area (from 13.36 to 35.53 m2/g) and crystallite size (from 200 nm to even 15 nm).

Raja et al. [36] compared the processing properties of starch, urea, and glycine in microwave combustion synthesis of Co3O4 nanopowders. Instead of traditional heating, they used microwave energy (2.45-GHz multimode cavity at 850 W for 10 min). The average crystallite sizes for the starch, urea, and glycine-assisted cobalt oxide nanopowders were 23.3, 26.9, and 30.2 nm, respectively. Another significant difference was in the shape formation of the obtained nanoparticles. Starch-assisted grains were nanorods, whereas urea- and glycine-assisted ones were spherical-like and nanospheres, respectively. In turn, magnetic measurements revealed that if the fuel is starch, cobalt oxide is ferromagnetic, whereas if the fuel is urea or glycine, cobalt oxide is superparamagnetic.

Visinescu et al. [37] used the low-temperature combustion-based synthesis to obtain zinc aluminate oxides. Starch was tested as a single fuel and in a two-fuel mixture with N-methylurea in various ratios. The combustion process started in the temperature range of 170–450 °C, but the pure spinel phase was obtained at 800 °C. It was proved that the presence of starch helped to lower the temperature and increased the time of combustion reaction, i.e., starch decreased the process exothermicity. Starch-derived zinc aluminate oxides were in the form of porous aggregates in contrast to N-methylurea-based denser forms.

To obtain BiFeO3, Koferstein [38] used a combustion method supported by starch that served as a complexing agent. (BiFe) gel was calcined at a temperature range of 500–750 °C, and pure-phase BiFeO3 powder was obtained already at 550 °C. Further increase in temperature caused gradual grain degradation to Bi25FeO40 and Bi3Fe4O9. This process was also dependent on the degree of heating and the fuel/oxidizer ratio.

Khattab et al. [39] used a modified combustion method to obtain Co3Mg1−xAl2O4 blue pigments. The output powders were prepared as a result of the microwave combustion method. Then, to optimize the spinel formation, the resulting powder precursors were calcined at 400, 550, and 800 °C. Starch used in this process as a combustion fuel caused the powder precursors to become oxidants. In addition, by analyzing the DTA and TG curves, it was found that the temperature degradation of starch contributed to the reduction in particle size of the powder.
The above studies confirmed excellent starch properties as combustion fuel among different organic derivatives such as urea and glycine. The more carbon in the precursor and the lower combustion enthalpy, the better properties of the ceramic powders synthesized by the combustion process can be obtained (Table 2). Due to the fact that starch can provide a lot of energy to the system during combustion, as a result of which the exothermcity of the reaction is reduced, the temperature at which the powder can be synthesized is also lowered. Despite these important advantages, there are a few things to keep in mind that can disturb the starch-assisted combustion process. It should be remembered about the mass ratio of starch to metal nitrate, i.e., fuel-to-oxidation ratio. If it will be lower or higher than the stoichiometric one, we will get impure powder or a fully nonsynthesized one.

### 2.3 Combined methods

Besides combustion method and sol–gel process, ceramic powders can be obtained by the use of other methods, mainly combined ones. Starch here plays a role of the capping agent and/or carbon source. For example, thanks to the combination of the sol–gel process and combustion synthesis, fine ceramic powders, mainly spinels, with high homogeneity and purity, can be obtained. The biggest group of the powders that can be obtained via the starch-assisted sol–gel combustion method are ferrites.

The starch-assisted sol–gel combustion method was used by Koferstein et al. [40] to obtain MgFe$_2$O$_4$, which was then subjected to isothermal sintering in the temperature range of 550–1100 °C for 1 or 10 h. The resulting sintered materials exhibited varying magnetic properties, depending on the particle size, which could be adjusted using starch. Yadav et al. [41] synthesized NiFe$_2$O$_4$ nanoparticles by the use of this method, where starch acted as the source of organic fuel. Gel obtained from the sol–gel process was heated up to 200 °C to initiate a self-sustaining combustion reaction. Such obtained powders were annealed in air atmosphere at 200–1000 °C. X-ray powder diffraction (XRD) analysis showed the improvement of crystallinity with the increment of annealing temperature. Also, the higher was the temperature, the bigger size of particles was observed, starting from 5 to 15 nm at 200 °C (spherical shape), up to 70–300 nm at 1000 °C (spherical and elongated shape). It was also proved that better crystallinity can be achieved by a proper ratio of starch and nitrate. They have used the same method to synthesize Co$_{1-x}$Zn$_x$Fe$_2$O$_4$ spinel ferrite nanoparticles [42], Nd$^{3+}$-doped CoFe$_2$O$_4$ spinel ferrite nanoparticles [43], and ZnFe$_2$O$_4$ spinel ferrite nanoparticles [44].

Dinh et al. [45] studied the effect of annealing temperatures (700, 800, and 900 °C) and Cu$^{2+}$/Fe$^{3+}$/starch ratio on the magnetic and catalytic properties of CuFe$_2$O$_4$ powders. The higher the annealing temperature was, the more tetragonal spinel structure phase was observed and the less impurity phases like CuO were observed. Increasing the starch content in the examined systems, it was found that it does not affect the morphology of samples, but the particle size (from 100 to even 500 nm) and the photocatalytic activity. The optimal annealing temperature was 800 °C and the optimal Cu$^{2+}$/Fe$^{3+}$/starch ratio was 1:2:3.

Ansari et al. [46] received CoTiO$_3$/CoFe$_2$O$_4$ nanopowders with enhanced magnetic and electrochemical properties via the sol–gel autocombustion method. They compared fuel, capping, and the reducing properties of glucose, maltose, and starch. The effect of added sugars, Ti-To-Co ratio, and calcination temperature was also studied. Gelation and autocombustion temperatures were 60 °C and 120 °C, respectively, whereas the calcination process was carried out at two temperatures (650 and 750 °C) and lasted for 1 h. It was proved that the more complex sugar used in the synthesis as a reducing agent, the better-formed powder grains can be obtained. Using glucose, a highly agglomerated product was obtained, whereas for maltose and starch aggregates, spherical nanoparticles were obtained. They also confirmed the fact that the more hydroxyl groups are in sugar, the better the powder grains are covered with carbohydrate and no large agglomerates are formed, and in the later stages, spherical powder particles are formed as a result of the combustion process. It is also easier to control the size of nanoparticles. When the calcination temperature was lower (more fuel in the system), the reaction rate also slowed down (particles had more time to react), which resulted in a more regular grain shape. It was also found that by reducing the amount of starch in the system, the grain size and degree of agglomeration increase.

Smirnova et al. [47] used a mixture of glycine and starch to synthesize Mg(Fe$_{0.8}$Ga$_{0.2}$)$_2$O$_4$ spinel nanopowders. The use of glycine alone causes contamination of the powder by dimensional inhomogeneity of particles and larger crystalites are formed (50–100 nm), whereas for the mass ratio of

### Table 2 Characterization of ceramic nanopowders obtained by the combustion process

| Synthesized powder | Temperature of the calcination [°C] | Crystallite size [nm] | References |
|--------------------|-----------------------------------|---------------------|------------|
| Al$_2$O$_3$–ZrO$_2$ | 500                               | 20                  | [32]       |
| AlN                | 1400                              | 100                 | [34]       |
| MgO                | 400                               | 8                   | [35]       |
| Co$_3$O$_4$        | –                                 | 23                  | [36]       |
| MgAl$_2$O$_4$      | 900                               | 17                  | [33]       |
| ZnAl$_2$O$_4$      | 800                               | 15–30               | [37]       |
| BiFeO$_3$          | 500–750                           | 36–243              | [38]       |
| Co$_{x}$Mg$_{1-x}$Al$_2$O$_4$ | 400–800 | 35–100              | [39]       |

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glycine to starch 8:1, pure Mg(Fe0.8Ga0.2)2O4 crystallites of 40–50 nm are formed.

In turn, Agilandeswari and Kumar [48], using starch-assisted sol–gel combustion method, obtained Ca3Co4O9 powder with improved magnetic and electrical properties. A sol–gel mixture, solution containing cobalt and calcium nitrates, and starch was used. By heating such a mixture to 120 °C, a gel, which after further thermal treatment converted into a noncarbonate Ca3Co4O9 phase, was obtained. On the basis of TG/DTA analysis, it was found that the starch presence contributed to a significant reduction in the reaction exothermicity, i.e., the exothermic oxidation of carbon generated a thermal reaction that allowed the conversion of cobalt oxide and calcium carbonate into calcium cobaltite.

Freitas et al. [49] compared four complexing agents: citric acid, glycine, corn starch, and gelatin in sol–gel autocombustion synthesis of LiCoO2 nanopowders. SEM and XRD research has shown the rhombohedral phase in all samples and a small amount of cubic Co3O4 spinel in the samples with starch and gelatin, as well as in the blank test (without any gelation additive). Gelation temperature of each sample varied between 70 and 80 °C, but the significant dissimilarity was in the gelation time. The longest time was for the samples based on citric acid (5 h), next gelatin and glycine (3 h), and the shortest time was for the samples based on starch (1 h). Combustion temperature was 300 °C and calcination temperature was 700 °C. Based on the XRD data, the calculated average size of LiCoO2 crystallites was as follows: 66–70 nm for glycine, 69–72 nm for citric acid, 70–75 nm for starch, and 73–75 nm for gelatin. The larger size of crystallites is related to the fuel type, because decomposition of starch and gelatin gives more energy to the system during the formation of nanocrystallites.

Motevalian and Salem [50] investigated the influence of glycine–potato starch mixing ratio on the structural properties of MgAl2O4 nanopowders synthesized with the use of sol–gel auto combustion method. Both additives are complexing agents as well as fuel additives. But starch changes ignition from flame to smoldering and because of that, it lowers the temperature of the autoignition process. Addition of starch counteracts agglomeration and facilitates the reaction between solid-phase and gas species, resulting in more pure spinel powder (purity increased after 10 wt% addition of starch).

Ramasami et al. [51] obtained photocatalytically and antimicrobially active ZnO powders via the sol–gel autocombustion process. As the carbohydrate additive, they have used tapioca starch pearls that they mixed with zinc nitrate hexahydrate in the mass ratio of nitrate:starch 1:1, 1:0.5, and 1:0.25. The best ratio was the middle one. Thanks to it, spherical nanoparticles of 45 nm were obtained.

Starch can also be a source of carbon in the synthesis of ceramic powders. Okano and Suzuki [52] developed a hydrothermal synthesis of carbonate spheres (CSs) using corn, potato, and rice starch. After hydrothermal treatment of aqueous starch suspensions at 160 °C for 4–16 h, followed by washing and drying, CSs were obtained. Chemically, these spheres did not differ from each other. As an effect of increasing starting mass and hydrothermal treatment time, larger particles were obtained. In the case of potato starch, probably due to its type B structure, the narrowest particles were obtained. This confirms the assumption that the initial crystalline structure of the starch interacts with hydrothermal reactions.

Wei et al. [53] obtained chromium oxide green pigment during the hydrothermal reaction in an autoclave (200 °C, 2.5 h) between Na2CrO4 and starch, and further calcination of amorphous Cr(OH)3 (950 °C, 2 h). The mineralizer effect (carbonates and urea) on the synthesis conditions of the pigment was also studied. The results indicate that conversion of chromate was promoted by the type of mineralizer and its content. The best results (98.9% of conversion) were obtained in the presence of NaHCO3 at 10–100 g/L, and when the temperature of the reaction was higher than 100 °C.

In turn, Yang et al. [54], in the single-stage carbonation-reduction process, where soluble starch served as a carbon carrier, received WC–Co composite nanopowders with spherical shape and average particle size in the range of 43–339 nm.

Visinescu et al. [55] obtained nanograins of Co1−xZnxAl2O4 (x = 0.1, 0.2, 0.4, 0.6, 0.8, and 1) blue pigment, using starch gelation process. In the first step, an aqueous solution containing starch and Zn (II), Co (II), and Al (III) salts was heated to 80 °C to obtain a gel. Then, the material was subjected to heat treatment at 800 °C to obtain a gel with high homogeneity of metallic precursors, which lowered the temperature required to spinel crystallization (from 713 to 684 °C).

As one can see, in combined methods, thanks to a synergy effect, starch allows to obtain ceramic powders with better properties and to lower the process temperature. Almost all types of ceramic powders (Table 3) can be obtained using combined methods, in particular, ferrites and spinels. In favor of starch is the fact that it gels faster than other organic agents. Due to the presence of a large number of hydroxyl groups in its structure, it is able to “stick” more tightly to the grains of the resulting powder, thanks to which a desirable shape of particles, generally spherical, with a low degree of agglomeration, is obtained. By controlling its quantity, grain size of the ceramic powders can also be controlled. However, it is important to choose a
proper fuel-to-oxidation ratio and not to overdo the starch content because we will get an impure final powder product.

### 3 Conclusions

The properties of ceramic powders largely affect the final properties of ceramic materials. The most important factors include high purity, the ability to control particle shape, homogeneity in chemical composition and phase, lack of aggregation, and narrow size range. Three groups of methods are used for the preparation of ceramic powders that are subjected to sintering: solid-phase synthesis, liquid-phase synthesis, and gas-phase synthesis. In this case, starch can act as a chelating and gelling agent (sol–gel method) or providing energy in the combustion synthesis process. Based on the shown examples, it can also be concluded that the use of starch allows to lower the temperature of the ceramic powder formation. We have proved that starch is an eco-friendly and easily transformable carbohydrate polymer that plays a differentiated role in the ceramic powder synthesis.

In the case of the sol–gel method, when the gel network is formed, starch helps to create a more compact and porous structure. Thanks to this, the resulting particles hang in space and do not agglomerate, and in the drying stage, when shrinkage and densification occur, more solvent is removed from the system. Despite the obvious advantages of the starch-assisted sol–gel process, only oxides and their mixtures can be synthesized using this method. In turn, considering the benefits of starch-assisted combustion method, the most important of them is the fact that starch is a highly energetic fuel that provides energy during combustion, and as a result, lowers the calcination temperature. Using this starch-assisted method, more types of ceramic powders, such as nitrates and spinels, in which synthesis requires a higher temperature, can be obtained. Among the combined methods, starch-assisted sol–gel combustion is the leader. A combination of the sol–gel process and combustion allows for even more accurate crystallite-size control. Thanks to this method, most types of ceramic powders can be synthesized, among which ferrites, mainly used for the production of magnetic materials, lead.

### Compliance with ethical standards

#### Conflict of interest

The authors declare that they have no conflict of interest.

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**Table 3** Characterization of ceramic nanopowders obtained by starch-assisted combined methods

| Synthesized powder | Type of process | Temperature of the calcination [°C] | Crystallite size [nm] | References |
|--------------------|----------------|------------------------------------|----------------------|------------|
| MgFe$_2$O$_4$      | Sol–gel autocombustion synthesis | 550–1100                          | 7–15                 | [40]       |
| NiFe$_2$O$_4$      |                            | 200–1000                           | 5–300                | [41]       |
| Co$_{1-x}$Zn$_x$Fe$_2$O$_4$ |                | 800–1900                           | ~19–40               | [42]       |
| Nd$^{3+}$-doped CoFe$_2$O$_4$ |            | 700–1000                           | ~4–20                | [43]       |
| ZnFe$_2$O$_4$      |                            | 400–800                            | 5–70                 | [44]       |
| CuFe$_2$O$_4$      |                            | 700–900                            | 100–500              | [45]       |
| CoTiO$_3$/CoFe$_2$O$_4$ |             | 650–750                            | 20–50                | [46]       |
| Mg(Fe$_{0.8}$Ga$_{0.2}$)$_2$O$_4$ |            | 700–1000                           | 40–50                | [47]       |
| Ca$_2$Co$_3$O$_4$  |                            | 700–1500                           | 150–300              | [48]       |
| LiCoO$_2$          |                            | 700–1750                           | 70–75                | [49]       |
| MgAl$_2$O$_4$      |                            | 900–1200                           | 12.5–19              | [50]       |
| ZnO                |                            | 500–700                            | 45–70                | [51]       |
| Carbonate spheres  | Hydrothermal synthesis     | 160–220                            | 0.22–7.20 µm         | [52]       |
| Cr$_2$O$_3$        |                            | 950–1100                           | ~                   | [53]       |
| WC–Co              | Single-stage carbonation-reduction process | 1100–1300 | 43–339              | [54]       |
| Co$_{1-x}$Zn$_{1-x}$Al$_2$O$_4$ | Starch gelation process | 800–1100                           | ~27                  | [55]       |
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