Writing and deleting skyrmions with electric fields in a multiferroic heterostructure

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Abstract

Magnetic skyrmions are topological spin textures that can be used as information carriers for the next-generation information storage and processing. The electric-field controlling of skyrmions in such devices is essential but remains technologically challenging. Here, using the first-principle calculation and the Ginzburg-Landau theory, we propose a reliable process for writing and deleting skyrmions by electric fields, on the platform of a multiferroic heterostructure, particularly the \( \text{Cr}_2\text{Ge}_2\text{Te}_6/\text{In}_2\text{Se}_3 \) heterostructure. We show that the electric field controls the electric polarization and indirectly influences the antisymmetric Dzyaloshinskii-Moriya interaction (DMI) between the magnetic moments. The latter is responsible for the generation and removal of the skyrmion spin textures, and we study this mechanism by the Ginzburg-Landau analysis. We discuss the real-space Berry curvature, topological Hall effects, possible quantum anomalous Hall effect, and other competing magnetic structures. These results represent examples of quantum technology and may have potential applications in future skyrmionics and the device fabrication.

Introduction.—In recent years, magnetic skyrmions have attracted much attention due to the spin textures and potential applications in the next-generation information storage and processing devices. Magnetic skyrmions are vortexlike topological objects in magnetic systems [1–7]. There has been a significant experimental evidence for their existence in condensed matter systems [8–12]. Each skyrmion is characterized by a topological invariant called the skyrmion number, which is an integer related to the homotopy mapping from the vector spin space to the real space and quantifies the winding of its spin configuration. Because of its topological nature, a continuous deformation of the spin configuration cannot vary the skyrmion number. Thus, magnetic skyrmions are rather robust objects that are topologically protected against environmental disturbance, and can be utilized as information carriers in future devices like the skyrmion-based racetrack memory [13].

In the potential devices, it is crucial to be able to write and delete the skyrmions. It was experimentally demonstrated that skyrmions can be created and destroyed by the tunneling current [14] or the electric field [15] of a scanning tunneling microscope (STM) tip, with the latter still involving a tunneling current. Another promising way is to control the skyrmions in the multiferroic insulators by electric fields via the electro-magnetic coupling [16–19]. This has the advantage of avoiding the inevitable energy dissipation in the current injection methods, and the spatial position of the skyrmion is less perturbed, which is better for a write unit [15]. However, the single-phase multiferroic materials are rare because ferromagnetism (ferroelectricity) arises from partially filled (empty) \( d \) shells of transition metal ions [20].

Since the discovery of graphene [21], technical advances have made feasible the fabrication of heterostructures from different van der Waals (vdW) materials [22] and provide rich degrees of freedom to form multifunctional materials. One crucial ingredient for skyrmions is the antisymmetric exchange interaction, known as Dzyaloshinskii-Moriya interaction (DMI) [23, 24] that originates from the spin-orbit coupling (SOC). DMI can only exist in systems without the inversion symmetry. Inversion symmetry breaking is an obvious property of a heterostructure; hence, there is usually a finite DMI for the magnetic interaction. Moreover, in 3D systems, the skyrmion lattice phase is usually restricted to a narrow region of temperature and external magnetic field. Only with the help of thermal fluctuations can the skyrmion lattice phase be stabilized against the competing conical phase [8]. In contrast, in 2D systems, the competing conical phase is absent for a perpendicular applied magnetic field, bringing about a much more robust skyrmion lattice phase that survives over a wide range of the phase diagram [9, 12, 25]. For these reasons, vdW heterostructures provide a versatile and natural platform for the exploration and application of magnetic skyrmions.

In this work, we study an electric controllable skyrmion lattice phase in the multiferroic vdW heterostructure made of ferromagnetic \( \text{Cr}_2\text{Ge}_2\text{Te}_6 \) and ferroelectric \( \text{In}_2\text{Se}_3 \). This is particularly inspired by an interesting recent work [27] by Gong et al., which proposed that this heterostructure can realize a switchable ferromagnet through the control of the electric polarization of \( \text{In}_2\text{Se}_3 \). By means of first-principles calculations, we find the existence of nonvanishing DMIs be-

![Fig. 1. (Color online.) (a) Top view of monolayer \( \text{Cr}_2\text{Ge}_2\text{Te}_6 \). (b) Side view of monolayer \( \text{Cr}_2\text{Ge}_2\text{Te}_6 \). (c) Top view of monolayer \( \text{In}_2\text{Se}_3 \). (d) Side view of monolayer \( \text{In}_2\text{Se}_3 \). The left and right panels show the structures with down and up electric polarization, respectively. Crystal structures are drawn by VESTA [26].](image-url)
tween the nearest-neighbor Cr\(^{3+}\) spins. The theoretical analysis based on the Ginzburg-Landau theory further predicts that in an appropriate external magnetic field, the skyrmion lattice phase is more stable than the ferromagnetic phase. More substantially, with the switch of the direction of electric polarization of In\(_2\)Se\(_3\), the strength of the DMI undergoes a change so significant that the magnetic structure of Cr\(_2\)Ge\(_2\)Te\(_6\) will switch between the topologically distinct skyrmion lattice and the ferromagnetic phase. This electric-field-controlled writing and deleting process of skyrmions should have potential applications in future skyrmionics devices.

Bulk Cr\(_2\)Ge\(_2\)Te\(_6\) is a layered material with the space group R\(\overline{3}\) [28]. Below 61 K, it develops a ferromagnetic order with Cr\(^{3+}\) moments aligned in the c axis. The long-range magnetic order has proved to survive in the 2D limit with an easy-axis anisotropy to counteract the thermal fluctuations [29]. The structure of monolayer Cr\(_2\)Ge\(_2\)Te\(_6\) is shown in Figs. 1(a) and 1(b). The magnetic Cr\(^{3+}\) ions arrange in a honeycomb lattice, as shown in Fig. 2(c), with each lattice site as a center of \(C_3\) rotational symmetry. Every nearest-neighbor Cr\(^{3+}\) ions are related by the inversion symmetry. Thus, the nearest-neighbor DMI is forbidden. To introduce nonvanishing DMIs, the inversion symmetry has to be broken. This is naturally achieved by manufacturing a heterostructure.

Bulk In\(_2\)Se\(_3\) is composed of Se-In-Se-In-Se quintuple layers stacking in the c direction [30]. In\(_2\)Se\(_3\) in few-layer forms have recently been obtained by mechanical exfoliation and chemical vapor deposition [31–33]. It was predicted that the ground state of the monolayer In\(_2\)Se\(_3\) does not have the middle Se layer equidistant from the two neighboring In layers [34]. Instead, it is nearer to either the upper or the lower In layer, leading to a spontaneous out-of-plane electric polarization whose direction depends on the middle Se layer position. Figures 1(c) and 1(d) depict the monolayer In\(_2\)Se\(_3\) with different polarization directions. The in-plane inversion symmetry is also broken, resulting in an additional in-plane electric polarization. These make In\(_2\)Se\(_3\) a 2D ferroelectric material, confirmed by later experiments [35, 36].

The heterostructure that we investigate is comprised of a monolayer Cr\(_2\)Ge\(_2\)Te\(_6\) and a monolayer In\(_2\)Se\(_3\), the same as Ref. [27]. The lattice constant of Cr\(_2\)Ge\(_2\)Te\(_6\) is fixed to the experimental value 6.83 Å. The \(\sqrt{3}\times \sqrt{3}\) In\(_2\)Se\(_3\) supercell is strained by -4.0% to fit with the Cr\(_2\)Ge\(_2\)Te\(_6\) primitive cell. Specifically, the interfacial Te atoms lie above the hollow sites of the interfacial In\(_2\)Se\(_3\) hexagon, as shown in Figs. 2(a) and 2(b). This is the configuration with the lowest energy [27]. A vacuum layer of 20 Å is introduced in the slab model to avoid the artificial interlayer interactions between the periodic images. The coordinates of the atoms inside the unit cell are relaxed.

The simulations are done within the framework of density functional theory [37, 38], using the plane wave basis sets and pseudopotential method as implemented in the Quantum ESPRESSO package [39, 40]. The energy cutoff for the plane wave basis is 72 Ry. The projector augmented-wave [41] pseudopotentials in the pslibrary [42, 43] are used.

The Brillouin zone is sampled by a \(6 \times 6 \times 1\) grid in structural optimizations. The convergence threshold for force is \(10^{-3}\) Ry/bohr. The Perdew-Burke-Ernzerhof exchange-correlation functional [44] is used together with Grimme’s D2 parameterization [45] of vDW correction to get a reliable interlayer distance. In the energy calculations, the local spin density approximation (LSDA) [46] is adopted. Fully relativistic pseudopotentials are employed to account for the SOC. On-site Hubbard U value is set to 0.5 eV for Cr \(d\) orbitals [29], which are orthogonalized using Löwdin’s method [47]. Four-state energy-mapping method [48, 49] is used to obtain the magnetic couplings. A \(1 \times \sqrt{3}\) supercell is adopted in the energy calculations to reduce the spin interaction with their periodic images, and the corresponding Brillouin zone is sampled by a \(8 \times 5 \times 1\) grid. The convergence threshold of total energy for self-consistency is \(10^{-7}\) Ry.

The optimized interlayer distances are 3.15 Å and 3.10 Å for the electric polarization of In\(_2\)Se\(_3\) pointing up and down, respectively. The total energy per unit cell with down polarization is lower than that with up polarization by 0.1 eV with SOC included and 0.06 eV without SOC. The magnetic moment of each Cr\(^{3+}\) ion is about \(3\mu_B\) for spin \(3/2\). These results are similar with the earlier report [27]. We adopt the generic Hamiltonian

\[
H_{12} = S_1 \cdot J \cdot S_2
\]

(1)

to describe the spin interaction between the Cr spins at site 1 and site 2 in Fig. 2(c). The exchange matrix has the components

\[
J = \begin{bmatrix}
J_{xx} & \Gamma_{xy} + D_z & \Gamma_{xz} - D_x \\
\Gamma_{xy} - D_z & J_{yy} & \Gamma_{yz} + D_x \\
\Gamma_{xz} + D_z & \Gamma_{yz} - D_x & J_{zz}
\end{bmatrix}
\]

(2)

in Cartesian coordinates. Here, the \(J\)'s are the Heisenberg interactions, the \(\Gamma\)'s are the off-diagonal pseudodipolar interactions, and the \(D\)'s are the DMIs. The Hamiltonian of other nearest-neighbor Cr sites can be deduced from \(H_{12}\). Our results of the parameters in Eq. (2) for normalized spin vector, \(|S| = 1\), are listed in Table I.

First, we analyze the case for a freestanding Cr\(_2\)Ge\(_2\)Te\(_6\). As mentioned earlier, the nearest-neighbor Cr sites are related...
by inversion symmetry, leading to $D = 0$. Moreover, there is an approximate symmetry of the mirror plane $\sigma_y$, perpendicular to the line joining the nearest-neighbor Cr sites, as Fig. 1(c) shows, although it is slightly broken [28]. In the following, we take this symmetry into consideration, and the point group becomes $C_{3v}$. Then, we have the symmetry restriction $\Gamma_{xy}, \Gamma_{yz} \approx 0$. Indeed, we find a vanishingly small $D_{xy}$ and $D_{yz}$, as can be seen from case 1 in Table I, validating our assumption.

When the monolayer Cr$_2$Ge$_2$Te$_6$ is placed on top of the ferroelectric monolayer In$_2$Se$_3$ to form a multiferroic heterostructure [27], the naturally broken inversion symmetry generates finite DMI. For the configuration shown in Fig. 2(a), the aforementioned approximate mirror plane symmetry $\sigma_y$ still holds and imposes the restriction $\Gamma_{xy}, \Gamma_{yz}, D_3 \approx 0$. Hence, the introduction of In$_2$Se$_3$ leads to nonzero $D_2$ and $D_4$, and this is supported by the numerical results in case 2 and case 3 of Table I. It is quite remarkable that the direction of the electric polarization of the substrate In$_2$Se$_3$ has a significant influence on the magnitude of $D_3$. We find $D_3$ is about four times larger with a downward electric polarization.

In contrast to the Heisenberg interactions, the DMI's necessarily favor noncollinear spin alignments. This is an important ingredient for the skyrmion formation. To discuss the possible skyrmion lattice phase, we employ the Ginzburg-Landau theory. For the $C_{3v}$ symmetry, the general free-energy functional of the spin distribution $S(r)$ under a magnetic field $B$ along $z$ direction is

$$F = \int d^2r \left\{ J_1 \left[ \partial_x S_x \right]^2 + \left[ \partial_y S_y \right]^2 \right\} + \frac{J_2}{2} \left[ \partial_x S_x \right]^2 + \left[ \partial_y S_y \right]^2 \right\} + \frac{J_3}{2} \left[ \partial_z S_z \right]^2 + \left[ \partial_y S_y \right]^2 \right\} + J_1 \partial_z S_x \partial_x S_y + \frac{\Gamma}{2} \left( \partial_x S_x \partial_y S_y - 2 \partial_y S_x \partial_y S_x - \partial_z S_z \partial_z S_z \right) + 2D \left( S_x \partial_x S_x + S_y \partial_y S_y + S_z \partial_z S_z \right) + \Delta S_x^2 + \Delta S_y^2 - BS_z \right\}.$$ (3)

Only terms linear and quadratic in $S(r)$ are considered at this stage. The parameters can be related to the microscopic ones in Table I by the Taylor expansion

$$S_\mu(r + d) \approx S_\mu(r) + d \cdot \nabla S_\mu(r) + \frac{1}{2} (d \cdot \nabla)^2 S_\mu(r),$$ (4)

for $\mu = x, y, z$. The results for the coefficients in the free energy are listed in the Appendix. By Fourier transformation

| Layouts   | $J_1$ | $J_2$ | $J_3$ | $\Gamma_1$ | $\Gamma_2$ | $\Gamma_3$ | $D_1$ | $D_2$ | $D_3$ |
|-----------|-------|-------|-------|------------|------------|------------|-------|-------|-------|
| (case 1) freestanding | -7.99 | -9.13 | -8.85 | 0.00       | 0.09       | 0.00       | 0.00  | 0.00  | 0.00  |
| (case 2) heterostructure (E $\uparrow$) | -11.41 | -12.55 | -12.31 | 0.00       | 0.13       | 0.00       | -0.05 | 0.00  | -0.36 |
| (case 3) heterostructure (E $\downarrow$) | -11.76 | -12.90 | -12.68 | 0.00       | 0.14       | 0.01       | -0.19 | 0.00  | -0.46 |

TABLE I. The calculated exchange couplings in Eq. (2), in units of meV. The spins are normalized with $|S| = 1$. The data for several different structures are listed: (1) freestanding monolayer Cr$_2$Ge$_2$Te$_6$; (2) Cr$_2$Ge$_2$Te$_6$/In$_2$Se$_3$ heterostructure with the upward electric polarization; and (3) Cr$_2$Ge$_2$Te$_6$/In$_2$Se$_3$ heterostructure with the downward electric polarization.

By Fourier transformation
by the second-order perturbation of $T'(q)$, that produces a $\theta$-dependent term $-\frac{1}{16}D^{2}F^{-1}(I_{2} - J)(I_{2} + J)^{-1} \cos 6\theta$. Therefore, the lowest eigenvalue of $T(q)$ will get its minimum value at six discrete $\theta_i, i = 1, \ldots, 6$. Having this six-fold degeneracy, we further construct an ansatz of the spin configuration from three degenerate spirals and a ferromagnetic $z$ component

$$S(r) = \frac{1}{\sqrt{2}} \sum_{i=1}^{6} \phi_{i} e_{i} \cdot e_{i} + \phi_{0} e_{z},$$

(7)

where the $e_{i}$'s are the eigenvectors in Eq. (6) for $\theta_i$, and the $\phi_{i}$'s are the corresponding order parameters. Moreover, we have $\theta_{2} = \theta_{1} + 2\pi/3, \theta_{3} = \theta_{1} + 4\pi/3, \phi_{i+3} = \phi_{i}$ for $i = 1, 2, 3$, and $\phi_{i+3} = \phi_{i}^{\ast}$. The ansatz is subject to the soft-spin-like constraint that the spatial average of the spin vector is normalized [51]

$$\langle |S(r)|^2 \rangle = \sum_{i=1}^{3} |\phi_{i}|^2 + |\phi_{0}|^2 = 1.$$  

(8)

We find that the free-energy density for this spin configuration is $f = \frac{1}{4}D^{2}F^{-1} \phi_{0}^2 - B\phi_{0} - \frac{1}{2}D^{2}F^{-1}$, where the wave vector $q = Q$ and the constraint in Eq. (8) are used. The free energy is minimized at $\phi_{0} = BJ D^{-2}$. However, there is still a degeneracy for different values of $\phi_{i}$ ($i = 1, 2, 3$) as long as the constraint Eq. (8) is honored. That is, the energy of a skyrmion lattice solution for which all the three $\phi_{i}$'s are equal in magnitude is degenerate with the single spiral solution for which only one of the three $\phi_{i}$'s are nonzero. This degeneracy remains after including the $T'(q)$ correction since it is still quadratic in $S(r)$.

The skyrmion lattice is at last stabilized by the quartic terms in the free energy [8]. For simplicity, we only include the leading isotropic term $\Delta F = \int \langle d^2 r | S(r)|^4 \rangle$. The coefficient, that is positive, is dropped because it does not matter in the following discussion. For the spin configuration in Eq. (7), the addition to the free-energy density due to the quartic term is

$$\Delta f = \phi_{0}^4 + \left(4\phi_{0}^2 + \sum_{i=1}^{3} \sum_{i=1}^{3} |\phi_{i}|^2 \right) \left(\sum_{i=1}^{3} |\phi_i|^2 \right)^2 + \frac{5}{8} \sum_{i=1}^{3} \sum_{j=1}^{3} |\phi_i|^2 |\phi_j|^2$$

$$+ \frac{9}{2} \phi_{0} \left( \phi_{1} \phi_{2} \phi_{3} + \phi_{1} \phi_{2} \phi_{3} \right),$$

(9)

For a skyrmion lattice solution, we have

$$\langle |\phi_{1}| = |\phi_{2}| = |\phi_{3}| = \left(\frac{1}{3} \left(1 - \phi_{0}^2\right)\right)^{1/2},$$

$$\Delta f_{\text{SKL}} = \phi_{0}^4 + \left(1 + 3\phi_{0}^2\right) \left(1 - \phi_{0}^2\right) + \frac{5}{12} \left(1 - \phi_{0}^2\right)^2$$

$$+ \frac{3}{12} \phi_{0} \left(1 - \phi_{0}^2\right)^{1/2} \cos (\alpha_{1} + \alpha_{2} + \alpha_{3})$$

(10)

where $\alpha_{i}$ is the phase of $\phi_{i}$ ($i = 1, 2, 3$). For a single-$q$ spin spiral, we instead have

$$\langle |\phi_{1}| = \left(1 - \phi_{0}^2\right)^{1/2}, \quad \phi_{2} = \phi_{3} = 0,$$

$$\Delta f_{\text{spiral}} = \phi_{0}^4 + \left(1 + 3\phi_{0}^2\right) \left(1 - \phi_{0}^2\right).$$

(11)

The condition $\Delta f_{\text{SKL}} < \Delta f_{\text{spiral}}$ demands that

$$\cos (\alpha_{1} + \alpha_{2} + \alpha_{3}) < - \frac{5}{12} \left(\phi_{0}^2 - 1\right)^{1/2}.$$  

(12)
magnetic (FM) and skyrmion lattice (SkX) phases, there is an energy barrier between them. The barrier of the most effective kinetic path between the two electric polarization directions was calculated to be 0.066 eV per unit cell [34]. There are experiments [35, 36] demonstrating that a bias voltage of several volts between the piezoresponse force microscopy tip and the In$_2$Se$_3$ thin film can switch the electric polarization direction. Therefore, the required electric field strength is accessible. The energy barrier between the FM and SkX phases and the spin dynamics related to the phase transition will be left to future works. Even if the barrier were high, we think the phase transition can be induced by a suitably applied magnetic field disturbance, which is superposed on the perpendicular magnetic field required by the stabilization of SkX phase. For example, Ref. [37] simulates the creation of skyrmions via the application of a tilted magnetic field pulse. The tilted magnetic field pulse excites spin waves in the FM phase, making the system energy higher than the barrier, so that the system will not get stuck in some metastable state. Then, the system can relax to the lower energy SkX phase. The transition from SkX to FM phase can also be induced by such disturbance.

Conclusions.—In summary, we have proposed an electric-field-controlled writing and deleting scheme of the magnetic skyrmions in the multiferroic vdW heterostructure Cr$_2$Ge$_2$Te$_6$/In$_2$Se$_3$ through electromagnetic coupling. The inversion symmetry breaking of the interface leads to nonvanishing DMIs between neighboring Cr local moments. Because of the introduction of the DMIs, a skyrmion lattice emerges in Cr$_2$Ge$_2$Te$_6$ with an appropriate magnetic field, and depends on the strength of the DMIs. The strength of the DMI is very sensitive to the direction of the electric polarization of the ferroelectric In$_2$Se$_3$, providing a natural scheme for the quantum controlling. Our calculations show that it is possible to create and destroy the skyrmion lattice phase by changing the direction of the electric polarization. Our findings may have a potential application in the future quantum technology such as the next-generation information storage and processing devices.

Note added.—Upon the acceptance of the current manuscript, we learned that a recent experimental work on multiferroic heterostructure consists of Pt/Co/Ta magnetic multilayer and ferroelectric Pb(Mg$_{1/3}$Nb$_{2/3}$)$_{0.7}$Ti$_{0.3}$O$_3$ was published in [58] where the identical phenomena were experimentally realized.

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Appendix A: The coefficients in the Ginzburg-Landau theory

The coefficients in the Ginzburg-Landau theory are given as

\[ \begin{align*}
\tilde{J}_1 &= -\frac{J_{xx} + 3J_{yy}}{4\sqrt{3}}, \\
\tilde{J}_2 &= -\frac{3J_{xx} + J_{yy}}{4\sqrt{3}}, \\
\tilde{J}_3 &= -\frac{J_{zz}}{\sqrt{3}}, \\
\tilde{\Gamma}_1 &= \frac{\Gamma_{zz}}{2\sqrt{3}}, \\
\tilde{D} &= \frac{D}{a}, \\
\tilde{A} &= \frac{\sqrt{3}}{a^2}(2J_{zz} - J_{xx} - J_{yy} + \frac{4}{3}A), \\
\tilde{C} &= \frac{\sqrt{3}}{a^2}(J_{xx} + J_{yy}).
\end{align*} \]

In \( \tilde{A} \), we have included a contribution from single ion anisotropy as follows

\[ T (q) = \tilde{C} I_{3x3} + \left[ \begin{array}{ccc}
\tilde{J}_1 q_x^2 + \tilde{J}_2 q_y^2 & -\tilde{J}_3 q_x q_y & \frac{\Gamma_1}{2} (q_x^2 - q_y^2) - i\tilde{D} q_x \\
-\tilde{J}_3 q_x q_y & \tilde{J}_1 q_y^2 + \tilde{J}_2 q_x^2 & -\Gamma_1 q_x q_y - i\tilde{D} q_y \\
\frac{\Gamma_1}{2} (q_x^2 - q_y^2) + i\tilde{D} q_x & -\Gamma_1 q_x q_y + i\tilde{D} q_y & \tilde{J}_3 q_x^2 + \tilde{J}_2 q_y^2 + \tilde{A}
\end{array} \right]. \]
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