High-quality $K_{0.47}Na_{0.53}NbO_3$ single crystal toward high performance transducer

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A large-sized, high-quality single crystal of $K_{0.47}Na_{0.53}NbO_3$ was grown by the top-seeded solution growth method. A complete set of elastic, dielectric, and piezoelectric constants for a [001]c-poled $K_{0.47}Na_{0.53}NbO_3$ single crystal was determined via combined resonance and ultrasonic methods. The [001]c-poled $K_{0.47}Na_{0.53}NbO_3$ (KNN47) single crystal exhibits a large piezoelectric coefficient ($d_{33} = 220 \text{ pC N}^{-1}$), high coupling coefficients ($k_{33} = 0.759$ and $k_t = 0.702$), high dielectric constants ($e_{11} = 2194$ and $e_{33} = 2162$), and relatively high elastic compliance constants ($s_{11}^{e} = 14.3 \times 10^{-12} \text{ N m}^{-2}$ and $s_{33}^{e} = 17.6 \times 10^{-12} \text{ N m}^{-2}$). Furthermore, a large electrical field induced strain of 0.16% was measured from the slope in the unipolar strain versus electric field dependence. The piezoelectric coefficient, $d_{33}$, was determined to be as high as 401 pm V$^{-1}$, likely resulting due to the morphotropic phase boundary around $x = 0.5$. These excellent properties make KNN47 a good candidate for application as a high-performance transducer material.

Introduction

Despite the favorable properties of lead-based piezoelectric materials, research on alternative lead-free piezoelectric materials has attracted a lot of attention over the last decade in order to mitigate concerns over the environmental and health effects of lead-based products in the atmosphere. Among these lead-free piezoelectric materials, potassium sodium niobate (K$_c$Na$_{1-x}$NbO$_3$, referred to as KNN$_x$) based materials are one of the most promising classes for real-world applications due to their relatively good piezoelectric response and high Curie temperature ($T_{c} = 415 \degree \text{C}$ of KNN50) compared with other lead-free materials, such as barium titanate. However, studying the nature of KNN materials in ceramics is more challenging than similar studies on single crystals because of the presence of grain boundaries in ceramics.

Among the pure KNN materials, the composition of potassium sodium niobate with $x = 0.5$ is the most interesting due to the presence of a turning point in the structural parameters, a phase transition from rhombohedral to orthorhombic and orthorhombic to tetragonal, and its density, remanent polarization, coercive field, and piezoelectric coefficient, $d_{33}$. Nearly all studies have typically focused on the monoclinic–orthorhombic morphotropic phase boundary in KNN with $x = 0.5$, which differs from the orthorhombic–tetragonal polymorphic phase transition in Li and Ta doped-KNN materials. The piezoelectric constant, $d_{33}$, of pure KNN crystal has been reported to be as high as 160 pC N$^{-1}$ with a high thickness electromechanical coupling coefficient, $k_y$, of 0.7 in large-sized [001]c-poled Li$_x$(Na$_{0.5}$K$_{0.5}$)$_{1-x}$NbO$_3$ ($x = 0.02$) single crystals. For both device design and fundamental studies, it is necessary to know the complete set of piezoelectric, dielectric and elastic properties of KNN single crystals. In order to obtain a large-sized single crystal with improved piezoelectric properties, recent years have seen significant efforts to develop more effective growth methods for larger KNN single crystals. Comparing with the flux, Bridgman and solid-state single crystal growth methods, it is more difficult to manipulate the crystal growth through controlling the temperature and growth rate. It is more dependent on experience, but ignoring the thermal stress and nucleation problems during the crystal growth. In a previous work, we have reported a complete set of matrix properties for domain engineered KNN80 single crystals. However, it has been found that high-quality, large single crystals of KNN50 are still challenging to grow due to difficulties in controlling the compositional uniformity, amongst other factors. There have been no reports of the piezoelectric, dielectric and elastic tensor properties in pure KNN crystal around the $x = 0.5$ boundary. The availability of a complete set of material constants for KNN80 single crystals motivates further studies to obtain a similarly complete set of data for KNN single crystals around $x = 0.5$.

In this study, a sufficiently large and high-quality single crystal of KNN47, with no cracks, was grown via the top-seeded solution growth (TSSG) method of large crucible with a mount.
of solution to grown relatively small crystal, as shown in Fig. 1. The dimensions of the KNN47 single crystal were $9 \times 9 \times 35$ mm$^3$. For the first time, we report a full property matrix set for the [001]$_C$-poled KNN47 single crystal from measured data around the $x = 0.5$ boundary.

**Experimental procedure**

Powders of K$_2$CO$_3$, Na$_2$CO$_3$, and Nb$_2$O$_5$ with purities of 99.99% were used as the raw materials and were mass-weighted to obtain the composition of KNN47; excess amounts of 10% (by mole) solutions of Na$_2$CO$_3$ and K$_2$CO$_3$ were added as a self-flux. The materials were mixed in a ball mill with ethanol for 24 h, and then placed into a Pt crucible and calcined at 850 °C for 6 h. The KNN47 single crystal was then grown at 1110 °C in a medium-frequency induction furnace under an air atmosphere. The velocities of rotation and pulling of the crystal were approximately 30 rpm and 4.0 mm per day, respectively. A about 5–6 days, the crystals were cooled down to room temperature at a rate of 0.5 °C min$^{-1}$. The crystal structures were confirmed by X-ray diffraction (XRD) (XRD-6000, Shimadzu, Kyoto, Japan), and the strain vs. electric field (S–E) loop of the crystals was recorded using a ferroelectric test system (Precision Premier II, Radiant Technology, Inc., Albuquerque, NM, USA).

For the effective macroscopic symmetry with the tetragonal point group 4 mm, there are a total of 11 independent material constants: six elastic constants, three piezoelectric constants, and two dielectric constants. All samples were oriented by a Laue X-ray machine with an accuracy of ±0.5 ° and cut from the same slice with uniform composition and aspect ratios.

Silver electrodes were painted on the two desired surfaces of each sample, and the samples were poled under an electric field of 100–200 V mm$^{-1}$ at 180 °C. Combined resonance/anti-resonance and ultrasonic methods were used to retrieve the 11 independent constants, allowing the determination of these coefficients with a high degree of self-consistency. In the resonance and anti-resonance frequency method, the electromechanical coupling factors, $k_{15}$, $k_{31}$, $k_{33}$, $k_{t}$, piezoelectric strain constants, $d_{31}$, $d_{33}$, and elastic compliance constants, $c_{11}^{\text{E}}$, $c_{33}^{\text{E}}$, $c_{44}^{\text{E}}$ ($=c_{55}^{\text{E}}$), $c_{11}^{\text{D}}$, $c_{33}^{\text{D}}$, $c_{44}^{\text{D}}$ ($=c_{55}^{\text{D}}$), and $c_{66}^{\text{D}}$ can be deduced from the phase velocities of the longitudinal and shear waves.

![Fig. 1](image1.png) Photograph of KNN47 single crystal grown via TSSG.

![Fig. 2](image2.png) (a) Temperature dependence of dielectric constants for heating and cooling processes. (b) X-ray powder diffraction patterns of the KNN47 single crystal.
Elastic stiffness constants: $C_{ij}^E$ and $C_{ij}^D$ ($10^{10}$ N m$^{-2}$)

| Material   | $C_{11}^E$ | $C_{12}^E$ | $C_{13}^E$ | $C_{33}^E$ | $C_{31}^E$ | $C_{33}^D$ | $C_{44}^E$ | $C_{66}^E$ | $C_{11}^D$ | $C_{12}^D$ | $C_{13}^D$ | $C_{23}^D$ | $C_{33}^D$ | $C_{44}^D$ | $C_{66}^D$ |
|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|
| KNN47      | 22.2       | 18.3       | 4.6        | 6.7        | 7.7        | 7.16       | 24.1       | 20.2       | 1.1        | 13.3       | 7.8        | 17.6       | 3.0        | 26.4       | 9.1        |
| KNN80      | 27.4       | 15.0       | 6.2        | 14.6       | 7.9        | 8.5        | 28.3       | 15.9       | 3.0        | 26.4       | 9.1        | 8.5        | 3.0        | 26.4       | 9.1        |
| KNNT       | 17.2       | 11.0       | 10.2       | 13.8       | 8.3        | 9.3        | 22.9       | 16.8       | 2.8        | 23.2       | 8.9        | 9.3        | 3.0        | 26.4       | 9.1        |

Elastic compliance constants: $s_{ij}^E$ and $s_{ij}^D$ ($10^{-12}$ N m$^{-2}$)

| Material   | $s_{11}^E$ | $s_{12}^E$ | $s_{13}^E$ | $s_{33}^E$ | $s_{31}^E$ | $s_{11}^D$ | $s_{12}^D$ | $s_{13}^D$ | $s_{33}^D$ | $s_{31}^D$ |
|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|------------|
| KNN47      | 14.3       | −11.4      | −2.0       | 17.6       | 13.0       | 14.0       | 14.0       | −11.7      | −0.2       | 7.5        |
| KNN80      | 5.4        | −2.7       | −1.2       | 7.9        | 12.7       | 12.0       | 5.2        | −2.9       | −0.3       | 3.9        |
| KNNT       | 11.9       | −4.3       | −5.6       | 15.5       | 12.0       | 10.7       | 9.4        | −6.8       | −0.3       | 4.4        |

Piezoelectric coefficients: $e_{ij}$ (C m$^{-2}$), $g_{ij}$ ($10^{-12}$ C N$^{-1}$), $f_{ij}$ (10$^{-3}$ Vm N$^{-1}$), and $h_{ij}$ (10$^6$ V m$^{-1}$)

| Material   | $e_{11}$ | $e_{31}$ | $e_{33}$ | $d_{31}$ | $d_{33}$ | $d_{11}$ | $g_{11}$ | $g_{31}$ | $g_{33}$ | $h_{11}$ | $h_{31}$ | $h_{33}$ |
|------------|----------|----------|----------|----------|----------|----------|----------|----------|----------|----------|----------|----------|
| KNN47      | 4.6      | −5.9     | 11.1     | 60       | −40      | 220      | 3.1      | −8.3     | 46       | 2.4      | −31.5    | 59.3     |
| KNN80      | 7.4      | −3.4     | 12.3     | 93.7     | −23.2    | 104.2    | 27.1     | −8.4     | 37.6     | 13.4     | −26.6    | 97.1     |
| KNNT       | 3.7      | −5.2     | 6.7      | 45       | −77      | 162      | 17.4     | −32.6    | 68.5     | 15.5     | −110.0   | 140.0    |

Relative dielectric constants: $\epsilon_{0}$, $\beta$ ($10^{-4}$ $\epsilon_{0}$$^{-1}$)

| Material   | $\epsilon_{11}^R$ | $\epsilon_{31}^R$ | $\epsilon_{33}^R$ | $\beta_{11}^R$ | $\beta_{31}^R$ | $\beta_{33}^R$ | $\beta_{13}^R$ | $\beta_{33}^R$ |
|------------|-------------------|-------------------|-------------------|----------------|----------------|----------------|----------------|----------------|
| KNN47      | 2194              | 540               | 2162              | 211.4          | 4.56           | 18.5           | 4.63           | 47.3           |
| KNN80      | 390               | 313               | 320               | 141            | 25.6           | 31.9           | 31.2           | 70.9           |
| KNNT       | 291               | 267               | 272               | 54             | 34.4           | 37.5           | 36.8           | 186            |

Electromechanical coupling coefficient: $k$

| Material   | $k_{15}$ | $k_{31}$ | $k_{33}$ | $k_{1}$ |
|------------|----------|----------|----------|---------|
| KNN47      | 0.120    | 0.151    | 0.759    | 0.702   |
| KNN80      | 0.45     | 0.19     | 0.706    | 0.67    |
| KNNT       | 0.234    | 0.46     | 0.827    | 0.646   |

$a \rho = 4.412 \times 10^3$ kg m$^{-3}$.

measured along [001]$_c$, [010]$_c$, [100]$_c$, respectively. The density of the samples was determined by applying Archimedes’ principle.

**Results and discussion**

Fig. 2a shows the temperature dependence of the dielectric constants of [001]$_c$-oriented KNN47 single crystals for heating and cooling processes. It can be seen that the dielectric constants exhibit two anomalies in both the heating and cooling process. The anomalies occurred at approximately 205 °C and 428 °C for the warming process and 186 °C and 425 °C for the cooling process. These values correspond to the orthorhombic–tetragonal phase transition temperature, $T_{O-T}$, and the tetragonal–cubic phase transition temperature, $T_{c}$, respectively. The differences between $T_{O-T}$ and $T_{c}$ for the heating and cooling processes are approximately 19 °C and 3 °C, respectively. The orthorhombic phase structure is demonstrated from the power XRD pattern of a KNN47 single crystal, exhibiting the typical perovskite structure of an orthorhombic phase, as shown in Fig. 2b. These data show that the KNN47 single crystals exhibit an orthorhombic phase structure at room temperature and, as $T_{O-T}$ is far from room temperature, the properties can be determined reliably.

The complete set of elastic, piezoelectric, and dielectric constants for the [001]$_c$-poled KNN47 single crystal are listed in Table 1, where they are compared to those of KNN80 and KNNT single crystals. The density was determined to be 4.412 × 10$^3$ kg m$^{-3}$. The $d_{33}$, $d_{15}$, and $d_{31}$ values, which were measured using the resonance method, are 220 pC N$^{-1}$, 40 pC N$^{-1}$, and 60 pC N$^{-1}$, respectively. Among them, $g_{11}$, $g_{31}$, and $g_{33}$ in KNN47 single crystals are over 2 times greater than those in KNN80 single
crystals, which strongly indicates a greater transverse mechanical adaptability along the [100]c and [001]c direction. The KNN47 single crystal also exhibits high transverse dielectric constants, with $\varepsilon_{11}^T = 2194$ and $\varepsilon_{11}^S = 2162$. These are much higher than the longitudinal dielectric constant, $\varepsilon_{13} = 540$, and the transverse dielectric constants seen in other materials.\(^{28-23}\) In comparison to the electromechanical coupling coefficient in KNN80 single crystal, $k_t$ and $k_{31}$ in KNN47 are significantly greater, while $k_{33}$ and $k_{15}$ are reduced. The impedance spectrum of electromechanical coupling coefficient, $k_t$, which is calculated from the formula:

$$k_t = \frac{\pi f_r}{2 f_a} \tan \left( \frac{\pi f - f_r}{2 f_a} \right),$$

where $f_r$ and $f_a$ represent the resonant and anti-resonant frequencies, is shown in Fig. 3.

Fig. 4 shows the dependence of the unipolar strain on the applied electric field of KNN47 single crystals, under a maximum electric field of 4 kV mm\(^{-1}\). The curve exhibits a small hysteresis with the measurement frequency of 1 Hz. The maximum strain of 0.16%, and corresponding piezoelectric coefficient, $d_{33}$, of 401 pm V\(^{-1}\), derived from the slope of the strain–field curve, were obtained when the electrical field reached 4 kV mm\(^{-1}\). The large value of $d_{33}$, which is much higher than that measured via the resonance method (220 pC N\(^{-1}\)) and Berlincourt method (250 pC N\(^{-1}\)), is likely due to the morphotropic phase boundary around $x = 0.5$. A brief comparison of the properties of KNN based materials is made in Table 2. The value of $d_{33}$ is much higher than those reported in ref. 24 and 25. At this time, the electromechanical coupling coefficient, $k_t = 0.702$, is the highest of any reported in the literature for KNN-based materials.

### Conclusion

In summary, we have grown a large-sized, high-quality KNN47 single crystal using the top seeded solution growth method. In order to better understand the properties of KNN47 single crystal, a complete set of elastic, dielectric, and piezoelectric constants for a [001]c-poled KNN47 single crystal has been determined via combined resonance and ultrasonic methods. The [001]c-poled KNN47 single crystal exhibits a large piezoelectric coefficient ($d_{33} = 220$ pC N\(^{-1}\)), which is the highest in pure KNN materials, high electromechanical coupling coefficients ($k_{33} = 0.759$ and $k_t = 0.702$, which are at the highest range), high dielectric constants ($\varepsilon_{11}^T = 2194$ and $\varepsilon_{11}^S = 2162$), and relatively high elastic compliance constants ($S_{11} = 14.3 \times 10^{-12}$ N m\(^{-2}\) and $S_{33} = 17.6 \times 10^{-12}$ N m\(^{-2}\)). Furthermore,

| Materials     | $d_{33}$ (pC N\(^{-1}\)) | $k_t$ (%) | $k_{33}$ (%) | $\varepsilon_{11}^T$ ($k_t$) [1 kHz] | $T_c$ (°C) | Ref.  |
|---------------|--------------------------|-----------|--------------|------------------------------------|------------|------|
| KNN47         | 220\(^a\), 250\(^b\), 401\(^c\) | 70.2      | 75.9         | 540                                | 428        | This work |
| KNN80         | 104.2\(^a\), 110\(^b\)    | 67        | 70.6         | 313                                | 455        | 20    |
| KNNT          | 162\(^a\), 200\(^b\)      | 64.6      | 82.7         | 267                                | 291        | 21    |
| Mn-doped KNN  | 161\(^a\)                | —         | 64           | 424 (10 kHz)                       | 407        | 26    |
| KNN           | 160\(^a\)                | 45        | —            | 240                                | 393        | 16    |
| 0.95KNN–0.05LiNbO\(_3\) | 405\(^a\) | 61        | —            | 185                                | 426        | 18    |
| KNN47         | 110\(^c\)                | —         | —            | 600                                | 230        | 24    |
| LF4T          | 416\(^a\), 750\(^c\)      | 61 ($k_t$) | —            | 1570                               | 253        | 5     |
| PZT4          | 410\(^a\), 700\(^c\)      | 60 ($k_t$) | —            | 2300                               | 250        | 5     |

\(^a\) Determined by resonance method. \(^b\) Measured by $d_{33}$ meter. \(^c\) Calculated from the slope of high field strain curve.
a large electrical field induced strain of 0.16% was measured. The piezoelectric coefficient, $d_{33}$, was measured to be as high as 401 pm V$^{-1}$, likely arising due to the morphotropic phase boundary around $x = 0.5$. These excellent properties make this KNN47 single crystal an ideal candidate for application as a high-performance transducer material.

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