In this paper bilayer thin films of 10CGO/8YSZ electrolyte with enhanced $O^2-$ conductivity composed of gadolinia doped CeO$_2$ (Ce$_{0.9}$Gd$_{0.1}$O$_{2-x}$) and yttria stabilized zirconia 8 mol% Y$_2$O$_3$ doped ZrO$_2$ layers were deposited by Pulsed Laser Deposition (PLD) using various numbers of pulses. This composite has supplementary function in electrochemical devices because 10 CGO can act as interdiffusion barrier with cathode materials. The structure and morphology of these thin films, having thicknesses smaller than 1µm, were investigated by X-ray diffraction (XRD), scanning electron microscopy (SEM) and atomic force microscopy (AFM). The XRD patterns exhibit cubic fluorite structure with a sharp (111) peak. The grain size was found to depend on the thickness. Optical characterization was carried out by spectroscopic ellipsometry (SE) using the Tauc-Lorentz model.

Keywords: Laser ablation, Electrolytes, Bilayer Thin Films, Tauc - Lorentz Model, Electrochemical Fuel Cells Devices

Thin films of high-k dielectric ceramics 10 CGO/8YSZ/Pt/Si with $O^2-$ conductivity deposited by Pulsed Laser Deposition [1] have been studied in this work for applications in miniaturized planar electrochemical devices, like ceramic multilayer oxygen lambda sensor and µSOFC (Micro Solid Oxide Fuel Cell) [2]. Unlike 8YSZ also 10CGO preserve fluorite structure over the whole range temperature from room temperature to the melting point at 2400°C. Due to the fact that performance is theoretically influenced by operating temperature and in order to overcome such constrains, it can be exploited the advantages of low resistivity and maximization of $O^2-$ conductivity of these thin films by doping ZrO$_2$ with 8 mol% Y$_2$O$_3$. Other methods include the development of composite bilayer thin films electrolytes 10CGO/8YSZ based on the higher $O^2-$ conductivity thin films. The CGO films ensure the higher ionic conductivity, most be dense and crack free necessary for high gas blockage and have very high mechanical strength to enlarge the time life of micro electrochemical devices; lower operating temperature (< 500°C) reduced the costs of materials and improves durability. In the low operating temperature, 10CGO thin films exhibit higher oxygen conductivity (one order of magnitude higher) than 8YSZ thin films because the ionic radius of Gd$^{3+}$ (0.105 nm) [5] is very close to the host cation Ce$^{3+}$ (0.108 nm). At the same time, the ionic transference of CGO [6] expands as the operating temperature decrease, being larger than 0.9 at T < 500°C. Our study is focused on developing bilayer electrolyte 10CGO/8YSZ thin films for operation in low temperature T < 500°C with CGO acting like a blocking layer for interdiffusion with cathode made of La,Sr$_{1−x}$Co$_x$Fe$_{1−y}$O$_{3−δ}$ (LSCF). In classical configuration with 8 YSZ thin films electrolyte perovskite like cathode react with YSZ leading to the formation of two insulating phases by segregation of Sr and La which destabilized the cathode electrolyte interference. To avoid a rapid deterioration of 8YSZ electrolyte, a dense 10CGO is deposited between La,Sr$_{1−x}$Co$_x$Fe$_{1−y}$O$_{3−δ}$ (LSCF) and YSZ to prevent a chemical reaction. But CGO electrolyte exposed to reduction atmosphere on anode side exhibit mixed ionic and electronic conductivity (MIEC) caused by reduction of Ce$^{4+}$ to Ce$^{3+}$ with influence on the global efficiency of electrochemical device by lowering the open circuit voltage and decreases the power output. For inhibiting chemical reaction of CGO and electrical leakage, YSZ can be placed on anode side. High density electrolyte 10CGO/8YSZ can be achieved regularly by sintering at T > 1400°C that generated larger grains, crakes and interdiffusion. Good quality of 10CGO, 8YSZ and 10CGO/8YSZ have been fabricated by techniques such as spin coating [8], spray pyrolysis [9], chemical solution deposition [10], dry press [11] and e-beam [12].

Pulsed Laser Deposition (PLD) is used for the deposition of novel thin films having a variety of functionalities and complex compositions, like 10CGO and 8YSZ, by optimizing the control parameters [1, 9, 13, 14]. PLD is the best technological research tool to prepare thin films having extremely pure phases because it allows a good stoichiometric transfer from target to substrate, without contamination. PLD can be used to grow planar ceramic multicomponent thin films and assure a good reproductibility and homogeneity.
Bilayer electrolyte 10CGO/8YSZ greatly reduces the ohmic and polarized resistance, and increased the electrochemical performance by blocking the diffusion of components from LSCF in the YSZ structure acting like a barrier layer. The microstructure of 10CGO/8YSZ thin films can be controlled by laser fluence, number of pulses, oxygen partial pressure, distance between target and substrate, substrate temperature. The thickness also influences the grain size on the rule $h \propto h^{2/5}$ [14]. The structural and morphological characterization of the thin films was carried out by XRD, SEM and AFM. Optical data is acquired by variable spectroscopic ellipsometry VWASE for investigation of the dielectric properties, roughness and thickness.

**Experimental part**

**Materials and methods**

The 10CGO target was synthesized from commercially Ce$_{0.90}$Gd$_{0.10}$O$_{1.95}$ nanocrystalline powder using a Pechini method. Ce (NO$_3$)$_3$$\cdot$6H$_2$O (Sigma-Aldrich) and Gd (NO$_3$)$_3$$\cdot$6H$_2$O (Sigma-Aldrich) like precursors, while citric acid (CA) (monohydrate, Sigma-Aldrich), and ethylene glycol (EG) (anhydrous, Sigma-Aldrich) were used as complexation and polymerization agents, respectively. The following molar ratios were used: CA:Me = 4.5 and CA:EG = 2. The removal of the excess solvent led to a gel. After gel drying overnight at 180°C, a solid resin was obtained. The solid resin was calcined in air at 550°C for 5 h and the mixed solid oxide was thus obtained. In order to prepare the target, the solid oxide powder was uniaxially pressed in a 10 mm diameter stainless steel die and the as-obtained pellet was further sintered at 1300°C for 10 h. The 8YSZ target with 14 mm diameter x 5mm thick was manufactured by American Elements (SUA) by sintering powder of 99, 99%Zirconia and 8mol% Yttria –Stabilized. The deposition of bilayer 10CGO/8YSZ/Pt/Si (111) thin films was performed on CompexPro 50 Pulsed Laser Deposition equipment using ArF excimer laser ($\lambda$ = 193 nm) with energy of 230mJ pulse$^{-1}$ and repetition rate $\nu$ = 30Hz, fluence $\varphi$ = 5 J/cm$^2$, in oxygen partial pressure of 8x10$^{-2}$mbar, the laser beam was focused at an angle of incidence of 45° on YSZ and CGO targets. Two sets of experiments were made by using Pt/Si (111) substrates (15 x 15mm) at 600°C and 50mm distance target-substrates with variation of number of pulses and time of depositions sample (a) (90.000; 50 min/ 90.000; 50 min) and sample (b) (70.000; 39min /140.000; 78min) for 10CGO/8YSZ/ Pt/Si bilayer configuration. The vacuum chamber has a base pressure 6x10$^{-4}$ mbar for sample (a) and 7 x 10$^{-4}$mbar for sample (b) by operating a turbo pump. (Fig.1)

Fig. 1. A schematic diagram of the pulse laser deposition setup

The XRD measurements have been performed by using a Panalytical X’Pert PRO MRD diffractometer equipped with monochromatic X-ray radiation of CuKα 1 (0.154056nm) in Bragg- Bretano geometry with 20 in the angular range 15-75°, scanned in steps of 0.02° with an acquisition time of 0.2s/step. The Joint Committee on Powder Diffraction Standards (JCPDS) database from the International Center for Diffraction Data (ICDD) was utilized for structural characterization like phase identification, crystalline dimensions and lattice constance.

The Scanning Electron Microscope (SEM) manufactured by FEI – SEM FEI Inspect S is designed to investigate the surface quality and cross – section for thickness measurements of bilayer 10CGO/8YSZ/ Pt/Si (111).

Atomic Force Microscopy (AFM) XE-100 Park Systems is used to characterization the morphology and Root Means Square (RMS) roughness, in non-contact mode; allows a maximum horizontal scan area of about 50 x 50 µm$^2$ and a vertical movement up to 12µm. The investigated areas were 2 x 2µm$^2$, 5 x 5 µm$^2$, 10 x 10 µm$^2$ and 20 x 20 µm$^2$, for revealing both the general aspect of the samples as well as their topographic details.
Spectroellipsometer WVASE with variable angels of incidence (60°-70°) is used for optical characterization, having high accuracy and precision with a wide spectral range 250-1700nm. Optical models a generated by WVASE32 software; n, k parameters and roughness are measured. The equipment is assisted by WVASE32 software necessary for generation the optical models and fitting the ellipsometers Psi (Ψ) and delta (Δ) parameters.

Results and discussions

The XRD characterization of 8YSZ and 10CGO ceramic targets is shown in Fig. 1. This is necessary to identify the crystalline structure that will be transfer on the substrate. Also, the characterization of the bilayer thin films of 10CGO/8YSZ/Pt/Si (100) is presented in relation to the same diagrams for identification of crystalline phases. The (111) crystalline phases are cubic phase, and are identified using the standard base XRD JSCPDS file 089-9069.

![XRD spectra of the two targets YSZ and CGO and bilayers CGO/ YSZ/Pt/Si (111) films. The diffractograms shows the films are crystalline and fluorite cubic structures](image)

The average crystallite sizes were calculated at the peak position of the (111) reflection (2θ = 40°), a strongly preferential orientation. The spectra indicate that the deposited films have polycrystalline structure, with lattice constants a = 0.5134 nm (YSZ) film (a) and a = 0.4923 nm for film (b) respectively. From the XRD pattern for films (b) it was observed also a reflection for (400) that indicated the polycrystalline structure of both samples. From Table 1 it is observed that the values of lattice constants for CGO/YSZ thin films and YSZ target are very close, due to the role of Y in stabilizing the cubic structure. The doping with Ce leads to a small difference in the mean dimension of particles, with an effect on crystalline order that will influence optical characterization (Table 1). The influence of the Pt/Si substrate is also appearing in Figure 2.

**Table 1**

| Sample and target | Substrate type | YSZ phase | CGO phase |
|-------------------|----------------|-----------|-----------|
|                   |                | Lattice parameter a (nm) | Crystallite dimension D (nm) | Lattice parameter a (nm) | Crystallite dimension D (nm) |
| YSZ Target        |                | 0.5155    | 18        | 0.54230    | 71         |
| CGO Target        |                | 0.5134    | 24        | 0.5426     | 41         |
| CGO/YSZ (a) P/Si  |                | 0.4923    | 20        | 0.5020     | 39         |
| CGO/YSZ (b) Pt/Si |                | 0.4923    | 20        | 0.5020     | 39         |
| ZrO2-cubic Standard JCPDS 089-9069 | 5,1350 | | | | |
In figure 2 show the surface morphology of the 10CGO/8YSZ film deposited on Pt/Si (111) substrates using various number of pulses and deposition times, generating variable RMS (Root Mean Square Roughness) of 34.6 nm and 19.2 nm, respectively, which are related to the PLD process parameters. Atomic Force Microscopy images [15] show fairly smooth surfaces and demonstrate that they are composed of nanoscale grains with distinct shape and particle distribution. The surface morphologies of the two bilayer films is changed with the local sites, dimensions of windows of exploration (2 x2 μm²), (5x5 μm²), (10x10 μm²) and (20x20 μm²), with a variation in RMS; this can be like a base for selecting PLD parameters for development more functional structures.

The surface morphology and cross section of the films are presented in figures 3a and 3b. From the top view it is observed the influence of the deposition parameters in obtaining stable thin films, free of cracks and small droplets; small particles have been obtained on sample (b). Cross- section micrographs indicate different thickness of 140.1nm for 10CGO and 510.0 nm for total thickness of sample (a) and 205.1nm for 10CGO and 615.0 nm for total thickness of sample (b). Both bilayers thin films are dense, columnar and crack free. Micrograph image obvious shows a better distribution of crystallites with influence of roughness and uniformity.
High-k ceramic electrolyte 8YSZ thin films are transparent over a large range of the spectrum; CeO$_2$ has a higher refractive index and high absorption mainly in ultraviolet (UV) radiation [16]. The substitution of Ce with Gd generates a significant effect on the coefficient of absorption [2, 17]. In the operating range (250-1700nm), optical characterizations were made using a VVASE (Variable Angle Spectroscopic Ellipsometry) (Woollam, SUA) assisted by the WVASE 32 software package for processing ellipsometric parameters $\Psi$ and $\Delta$ acquired at three angles of incidence (60, 65 and 70$^\circ$) based on an optical model [16]. Because 10CGO/8YSZ/Pt/Si produces a different absorption caused by interdiffusion effects, crystallite size and microstructure, Cauchy - Urbach laws are not sufficient to describe the model. As a result, an advanced Tauc - Lorentz dispersion model based on general oscillator (GenOsc) was selected to an accurate representation of absorption feature bellow the band gap, very important aspect for sensor design [18,19]. These aspects demonstrate the complexity of our bilayer structures.

Fig. 5. Variation of refractive and extinction coefficients as a function of ($\lambda$) for 10CGO/8YSZ/ Pt/Si thin films for Tauc Lorentz oscillator

In table 2 it is presented the roughness measured by SE and RMS measured by AFM like a comparative study. The difference in measurements is generated by the fundamental principles of SE and AFM. SE measures the effect of roughness at the level of atomic dimensions; RMS roughness is influenced of the peak to valley heights being about two times larger than SE measurements.

| Sample | Deposition temperature (°C) | Roughness measured by SE (nm) | RMS roughness AFM (nm) | Thickness nonuniformity (%) | MSE |
|--------|-----------------------------|------------------------------|------------------------|-----------------------------|-----|
|        |                             | 2x2$\mu$m$^2$ | 5x5$\mu$m$^2$ | 10x10$\mu$m$^2$ | 20x20$\mu$m$^2$ |     |
| Sample A | 600                         | 5.257            | 29.3              | 34.6              | 32.825         | 81.156 | 8.452 | 435 |
| Sample B | 600                         | 26.809           | 19.231            | 19.054            | 20.856         | 22.208 | 60.464 | 525 |

Conclusions

In this report it was investigated structural and optical characterization of the bilayer 10CGO/8YSZ/ Pt/Si thin films deposited by PLD with variable number of pulses. The aim is to use PLD like a reproductible technique suited for application in manufacturing planar ceramics micro electrochemical devices like $\mu$SOFC and $\lambda$ oxygen sensors. By tuning control parameters for PLD it was possible to deposit dense columnar, free cracks and voids bilayer 10CGO/8YSZ/ Pt/Si thin films as desired for solid electrolytes.
A novelty is the implementation of a GenOsc (Tauc Lorentz) optical model necessary for a better calculation of absorption mainly in UV spectra.

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References
1. B. ARNDT, H. NOEL, T. F. KELLER, P. MÜLLER, V. VONK, A. NENNING, A. K. OPITZ, J. FLEIG, U. RÜTT, A. STIERLE, Thin Solid Films, 603, 2016, p. 56-61
2. G. CONSTANTIN, C. ROSSIGNOL, J.P. BARNES, E. DJURADO, SOLID STATE IONICS, 235, 2013, p. 36-41
3. W. WU, Z. ZHAO, X. ZHANG, Z. LIU, D. CUI, B. TU, D. OU, Electrochemistry Communications 71, 2016, p. 43-47
4. M. Z. KHAN, M. T. MEHRAN, R. H. SONG, J.W. LEE, S. B. LEE, T. H. LIM, S. J. PARK, Ceramics International 42, 2016, p. 6978-6984
5. B. MORENO, R. F. GONZALEZ, J. R. JURADO, A. MAKRADI, P. NUNEZ, E. CHINARRO, International Journal of Hydrogen Energy 39, 2014, p. 5433-5439
6. M. ZUBAIRKHAN, M. T. MEHRAN, R. H. SONG, J.W. LEEA, S. B. LEE, T. H. LIM, S. J. PARK, Ceramics International 42, 2016, p. 5433-5439
7. T. MUKAI, S. TSUKUL, K. YOSHIDA, S. YAMAGUCHI, R. HATAYAMA, M. ADACHI, H. ISHIBASHI, Y. KAKEHI, K. SATOH, T. KUSAKA, K. C. GORETTA, Journal of Fuel Cell Science and Technology, 10, 2013, p. 061006-6
8. P. PLONCZAK, M. JOOST, J. HJELM, M. SØGAARD, M. LUNDBERG, P. V. HENDRIKSEN, Journal of Power Sources 196, 2011, p. 156-1162
9. B. SCHERRER, J. MARTYNCZUK, H. GALINSKI, J. G. GROLIG, S. BINDER, A. B. HÜTTER, J. L. M. RUPP, M. PRESTAT, L. J. GAUCKLER, Advanced Functional Materials, 22, 2012, p. 653-660
10. E. O. OH, C. M. WHANG, Y. R. LEE, S. Y. PARK, D. H. PRASAD, K. J. YOON, B. K. KIMC, J. W. SON, J. H. LEE, H.W. LEE, Ceramics International 40, 2014, p. 8135-8142
11. H. CHOI, G. Y. CHO, S.W. CHA, International Journal of Precision Engineering and Manufacturing-Green Technology, 2, 2014, p. 95-99
12. G. LAUKAITIS, D. VIRBUKAS, Solid State Ionics 247, 2013, p. 41-47
13. M. MISHRA, P. KUPPUSAMI, S. RAMYA, V. GANESAN, A. SINGH, R. THIRUMURUGESAN, E. MOHANDAS, Surface & Coatings Technology 262, 2015, p.56-63
14. N. PRYDS, K. RODRIGO, S. LINDEROTH, J. SCHOU, Appl. Surf. Sci. 255, 2009, p. 5232-5235
15. K. MUTHUKKUMARAN, P. KUPPUSAMI, T. MATHEWS, E. MOHANDAS, S. SELADURAI, Materials Science-Poland 3, 2007, p. 671-678
16. H. FUJIWARA, John Wiley &Sons, 2007, p.170
17. L. GAO, F. LEMARCHAND, M. LEQUIME, Thin Solid Films 520, 2011, p. 501-509
18. D.V. LIKHACHEV, N. MALKOVA, L. POSLAVSKY, Thin Solid Films 589, 2015, p. 844-851
19. D.FRANTA, M. CERMÁK, J. VOHÁNKA, I.OHLÍDAL, Thin Solid Films 631, 2017, p.12-22

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