PdAg Nanoparticles within Core-Shell Structured Zeolitic Imidazolate Framework as a Dual Catalyst for Formic Acid-based Hydrogen Storage/Production

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Formic acid (FA; HCOOH) is one of the most promising candidates for the storage of hydrogen (H₂). Herein, we report a H₂ storage/production system based on the hydrogenation of CO₂ and dehydrogenation of FA, using a nanostructured heterogeneous catalyst. Pd¹Ag² nanoparticles with an average size of 2.8 nm were encapsulated within a zeolitic imidazolate framework (ZIF-8) having a core-shell structure (ZIF-8@Pd₁Ag₂@ZIF-8). This composite displayed high activity and stability during both the hydrogenation of CO₂ to produce FA and the dehydrogenation of FA into H₂. This improved performance is attributed to the use of ultrafine Pd₁Ag₂ nanoparticles as well as the spatial regulation of the nanoparticles within the reaction field. This study suggests a new strategy for controlling the spatial distribution of metal nanoparticles within MOFs so as to fine-tune the catalytic activity and selectivity of ZIF-8@metal nanoparticles@ZIF-8 catalysts.

Hydrogen (H₂) is considered to represent a clean, nontoxic, sustainable and high-density energy carrier. However, the safe and economical storage, handling and transportation of H₂ remains a challenge. FA (FA; HCOOH) is a promising H₂ storage liquid, and has the advantages of low toxicity and a high hydrogen content (4.4 wt%), and so has received tremendous research interest. H₂ can be controllably released through the dehydrogenation of FA (ΔG = −32.8 kJ mol⁻¹) at room temperature in the presence of a suitable catalyst. CO₂-FA mediated H₂ storage systems have been proposed. The direct hydrogenation of CO₂ to FA over a catalyst is considered to be an economical, environmentally friendly method of using FA to store H₂, and benefits from the use of CO₂ as a feedstock. Therefore, there is currently a search for catalysts that promote both the hydrogenation of CO₂ and the dehydrogenation of FA.

Recently, homogeneous metal complexes have proven to be efficient catalysts for both CO₂ hydrogenation and FA dehydrogenation. It has also been demonstrated that electron-rich metal centers can significantly enhance the catalytic activities of such materials. Numerous studies have thus been dedicated to the fabrication of electron-rich active metal centers using electron-donating ligands, and high catalytic activities and selectivities have been reported. Despite such achievements, these catalysts remain difficult to recover and may exhibit poor stability, as well as the need for the concurrent use of organic solvents and harsh reaction conditions. Thus, it would be preferable to use heterogeneous catalysts for hydrogenation of CO₂ to FA or for FA decomposition because these materials are easier to handle and to recover. Recent studies have demonstrated that Pd nanoparticles represent one of the most active catalysts for CO₂ hydrogenation to formate and for FA decomposition, affording considerable yields in the presence of additives. The surface electron density of Pd nanoparticles has
also been shown to significantly affect their catalytic activity. Alloying Pd with other metals having different work functions so as to tailor the surface electron density of the Pd is emerging as a promising strategy for tuning the electron density of the Pd. PdAu, PdAg, PdCu, PdAuCo, and PdCuCr have all been synthesized and have shown enhanced catalytic activities as compared to monometallic Pd. In the present study, Ag was alloyed with Pd due to the low cost and high ductility of the former metal. Furthermore, considering that the electronegativities of Pd and Ag are 2.20 and 1.90, respectively, electron density should readily transfer from Ag atoms to Pd atoms, leading to the formation of electron-rich Pd, which plays an important role in CO2 hydrogenation as well as FA dehydrogenation.

In general, metal nanoparticles can offer high surface-to-volume ratios and abundant active sites, as well as significantly enhanced activities as compared to larger particles. However, these nanoparticles tend to have high surface energies and readily aggregate during catalytic reactions, especially upon heating, which significantly reduces their most helpful properties. Consequently, considerable effort has been applied to the stabilization of metal nanoparticles. One promising strategy is to encapsulate metal nanoparticles inside metal organic frameworks (MOFs). MOFs, a class of porous coordination polymers, are built using metal ions as connecting centers and organic molecules as linkers, and have emerged as an attractive class of functional materials widely used in catalysis because of their high surface areas, increased porosities, permanent nanoscale cavities or open channels, and chemical diversity. The use of MOFs as supports for metal nanoparticle catalysts should prevent agglomeration and detachment of the nanoparticles, thus preserving their intriguing properties in catalysis applications. There are two main approaches to incorporating metal nanoparticles in MOFs. The first, and most popular, method is the so-called "ship in a bottle" technique, which involves the absorption of a metal precursor into previously formed porous materials followed by the reduction of the precursor to give metal nanoparticles. However, metal nanoparticles having a broad size range and unpredictable spatial distribution are inevitably formed on the external surfaces of the MOF. Another approach is termed "bottle around ship", and consists of the synthesis of individual surfactant-stabilized metal nanoparticles that are subsequently coated with the MOF. Although remarkable achievements have been reported in this field, a common and facile strategy to encapsulate metal nanoparticles in MOFs in conjunction with controllable spatial distributions and small particle sizes is still lacking.

The zeolitic imidazolate framework (ZIF-8), having a zeolite-type structure with large cavities and small apertures, is well known for its chemical robustness and thermal stability, and can be synthesized at room temperature. It has also been reported that metal nanoparticles supported on ZIF-8 show excellent catalytic activity and selectivity during FA dehydrogenation, possibly due to the presence of soft Lewis acid sites and functional groups capable of activating the FA. To mitigate the aggregation of metal nanoparticles on the external surfaces of MOFs, as well as to prevent damage to MOFs during the post-reduction process, the present study developed a simple means of controllably encapsulating metal nanoparticles within ZIF-8. This "bottle around ship" approach involves the growth of a ZIF-8 core during an initial stage, using 2-methylimidazole (Hmin) moieties as organic linkers and Zn2+ ions as connecting centers, followed by the loading of PVP-stabilized PdAg nanoparticles onto the external surfaces of the ZIF-8 core, then coating of the nanoparticles with additional ZIF-8. Figure 1 illustrates the synthetic route for the controllable fabrication of the ZIF-8@PdAg@ZIF-8 catalyst. The composition and location of the incorporated PdAg alloy nanoparticles can be readily controlled via this process. The as-prepared ZIF-8@PdAg@ZIF-8 composite was examined as a catalyst for use in a hydrogen storage/production system based on FA. The PdAg alloy nanoparticles when confined by the crystallization process of the ZIF-8 afford a novel ZIF-8@PdAg@ZIF-8 catalyst that exhibits high activity and superior stability during the hydrogenation of CO2 to FA and the dehydrogenation of FA to CO2 and H2.
Results and Discussion

Bimetallic Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles were successfully synthesized using a non-aqueous method. As shown in Fig. S1, well-dispersed Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles were obtained. These nanoparticles displayed a very narrow size distribution, as demonstrated by Fig. S1b, and no agglomeration was observed. The average diameter was determined to be 2.2 nm. The alloy structure was assessed by examining the chemical environments of the Pd and Ag species by spectroscopic methods. Figure 2A displays the Fourier transform (FT) of $k^3$-weighted extended X-ray absorption fine structure (EXAFS) data at the Pd K-edge of the Pd\textsubscript{1}Ag\textsubscript{2} solution, as well as results for reference samples. In the case of Pd foil, the adjacent Pd–Pd bonds in the metallic form generated a single peak at approximately 2.5 Å, while two peaks were produced by the PdO, ascribed to the Pd–O shell and the Pd–O–Pd shell\textsuperscript{37}. Similar to the Pd foil, the Pd \textsubscript{1}Ag solution and the ZIF-8@Pd\textsubscript{1}Ag\textsubscript{2}@ZIF-8 generated a main peak corresponded to metallic Pd–Pd bonding. However, the Pd–Pd distance was slightly longer as compared to the foil, demonstrating the formation of heteroatomic bonding in the Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles\textsuperscript{20}. In the case of the Ag K-edge FT-EXAFS spectra, the Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles, ZIF-8@Pd\textsubscript{1}Ag\textsubscript{2}@ZIF-8 and Ag foil also showed a single peak at approximately 2.67 Å, and the interatomic distance in the Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles encapsulated in the ZIF-8 was shorter than that in the Ag foil. Thus, the formation of heteroatomic Pd–Ag bonding was confirmed (Fig. 2B)\textsuperscript{20}. In addition, a peak corresponding to the Ag–O bond (such as that generated by AgO at approximately 1.7 Å) is absent. On the basis of the above results, it is reasonable to assume that the Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles had an alloy structure.

The coordination numbers (CNs) and bond lengths ($R$) of the ZIF-8@Pd\textsubscript{1}Ag\textsubscript{2}@ZIF-8 sample as determined from EXAFS curve fitting are presented in Table S1. The data were well fitted using not only Pd–Pd and Ag–Ag bonds but also heteroatomic Pd–Ag bonds. The Pd–Pd bond distances were evidently shorter than those of Pd–Ag bonds, which in turn were shorter than Ag–Ag bonds. These results are consistent with the shifts observed in the main peaks of the FT-EXAFS spectra due to metallic bonding. It is widely accepted that the atoms inside an fcc lattice have a CN of 12. The average diameter of the encapsulated PdAg was 2.8 nm whose total CN is less than 12. Because the proportion of Pd in the nanoparticles was less than that of Ag (Pd:Ag = 1:2), the CN\textsubscript{total} (CN\textsubscript{Pd–Pd} + CN\textsubscript{Ag–Ag}) at the Pd K-edge was lower than that (CN\textsubscript{Ag–Ag} + CN\textsubscript{Pd–Ag}) at the Ag K-edge.

Crystallographic information regarding the samples was obtained by X-ray diffraction (XRD) analysis, with the results shown in Fig. 2C. The diffraction pattern of the pure ZIF-8 is in good agreement with a reported simulated pattern, demonstrating the successful fabrication of this material\textsuperscript{38}. The ZIF-8@Pd\textsubscript{1}Ag\textsubscript{2}@ZIF-8 sample also generated a diffraction pattern similar to that of the pure ZIF-8, suggesting that the encapsulation of Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles within the ZIF-8 did not change the framework structure. However, the intensities of the diffraction peaks were weaker than those of the pure ZIF-8, presumably because the encapsulated Pd\textsubscript{1}Ag\textsubscript{2} nanoparticles introduced some disorder into the MOF crystal. In addition, diffraction peaks assignable to Pd and Ag do not appear in the catalyst pattern, possibly because of the low concentrations of these elements and the small particle size. The pore structures of the samples were characterized using the N$_2$ sorption technique. As shown in Fig. 2D, the pure ZIF-8 produced a type-I isotherm that was completely reversible, which is typical of microporous materials. The Brunauer-Emmett-Teller (BET) surface area determined by N$_2$ adsorption-desorption for this material
was 1110 m² g⁻¹. The Pd₁Ag₂ nanoparticles encapsulated in ZIF-8 produced a similar isotherm to that of pure ZIF-8 except for a slight decrease in the N₂ uptake, suggesting a decrease in the number of micropores after the encapsulation of the Pd₁Ag₂ nanoparticles. This resulted in a slight decrease in the surface area to 926.3 m² g⁻¹. Based on the XRD and BET analyses, it appears that the crystallinity and porosity of the ZIF-8 are well preserved after Pd₁Ag₂ encapsulation.

The morphologies of the pure ZIF-8 and the ZIF-8@Pd₁Ag₂@ZIF-8 are shown in Fig. 3. The pure ZIF-8 had a rhombic dodecahedral morphology in conjunction with a particle size of approximately 350 nm. The rhombic dodecahedra were also uniformly dispersed without any significant aggregation. The morphology of the ZIF-8@Pd₁Ag₂@ZIF-8 did not undergo any obvious changes from that of the pure ZIF-8. The size and spatial distribution of the Pd₁Ag₂ nanoparticles were assessed by transmission electron microscopy (TEM), and the results are summarized in Fig. 4. Pd₁Ag₂ nanoparticles covered by a thin shell of ZIF-8 can be clearly observed, with the shell having a thickness of approximately 5 nm, as shown in Figs 4A and S2. The average size of the encapsulated Pd₁Ag₂ nanoparticles was determined to be 2.8 nm and these nanoparticles had a very narrow size distribution (Fig. 4B). In addition, no Pd₁Ag₂ nanoparticles were deposited on the external surface of the ZIF-8@Pd₁Ag₂@ZIF-8. The high-resolution TEM (HRTEM) image clearly shows the (111) and (20-1) planes, with lattice spacings of 2.31 and 1.77 Å (Fig. 4C,D), respectively. It should be noted that the lattice spacing of the (111) plane of Pd₁Ag₂ is smaller than that of the (111) plane of Ag, but larger than that of the (111) plane of Pd, while the lattice spacing of the (20-1) plane of Pd₁Ag₂ is between those of the (20-1) planes of Pd and Ag (see Table S2 for details). These results provide further evidence that the Pd₁Ag₂ nanoparticles had a true alloy structure.

The ZIF-8@PdAg@ZIF-8 catalyst was subsequently applied to the hydrogenation of CO₂ to produce FA. The reaction was carried out in a stainless steel reactor containing a 1 M aqueous NaHCO₃ solution (10 mL) at
100 °C for 24 h under a total pressure of 2.0 MPa (H2:CO2 = 1:1). Various catalysts were prepared by varying the molar ratio of Ag to Pd from 0:3 to 2.5:0.5 to determine the optimal composition, and Fig. 5 summarizes the catalytic activities over the different catalysts. No reaction occurred on the unsupported Pd1Ag2 nanoparticles when using the same amount of Pd1Ag2 as was encapsulated within the ZIF-8. This negative result occurred because the unsupported Pd1Ag2 nanoparticles aggregated into larger particles under the catalytic reaction conditions, as evidenced by the TEM image after the reaction (Fig. S3); the surface energy increases as the particle size decreases, which frequently leads to serious aggregation of the ultra-fine particles in order to minimize the total surface energy39. This phenomenon demonstrates the importance of ZIF-8 in preventing the aggregation of Pd1Ag2 nanoparticles during the catalytic reaction. On the other hand, we can still observe the small amount of colloidal Pd1Ag2 nanoparticles with a mean diameter of ca. 2 nm. This result suggests that stabilization of PdAg nanoparticles with strongly binding PVP ligands was inactive because of the prevention of the interaction with reactants40. It is interesting to observe that, in each case, Pd1Ag2 nanoparticles associated with ZIF-8 exhibited higher catalytic activity than that of ZIF-8@Pd3@ZIF-8, demonstrating that the PdAg alloy nanoparticles have a positive effect on the catalytic reaction. The catalytic activities of the various ZIF-8@PdAg@ZIF-8 samples were also highly dependent on the composition of the PdAg nanoparticles, and the optimal Pd:Ag ratio was found to be 1:2. The ZIF-8@Pd1Ag2@ZIF-8 thus exhibited the highest catalytic activity among all samples, with a value of 16.68 mmol g(catal.)−1 after 24 h. This level of activity was almost twice that obtained over ZIF-8@Pd3@ZIF-8 under the identical reaction conditions. Thus, decreasing the Pd:Ag ratio from 3:0 to 1:2 increased the catalytic activity. It has been widely reported that electron-rich metal centers by alloying with Ag can significantly enhance catalytic activity during FA dehydrogenation20,41. However, further decreases in the Pd:Ag ratio lowered the catalytic activity due to a decrease in the number of active Pd sites.

As noted, charge was transferred from the Ag atoms to the Pd atoms because the electronegativities of Pd and Ag are 2.20 and 1.9, respectively. Density functional theory (DFT) calculations employing Pd11 and Pd11Ag11 clusters as models of monometallic and alloy nanoparticles confirmed that the Pd atoms in the Pd11Ag11 clusters were indeed negatively charged in comparison with those in a Pd11 cluster, while the Ag atoms were positively charged as a result of charge transfer (Fig. 6). Additionally, the DFT results demonstrated that the highest occupied molecular orbital (HOMO) of the Pd11Ag11 cluster had a greater energy level than that of the monometallic Pd22 cluster, with values of −3.85 and −3.71 eV, respectively. This elevated HOMO level would be expected to increase the electron-richness of the active Pd sites.

Figure 7A displays a proposed mechanism for CO2 hydrogenation, initiated by the dissociation of H2 at a Pd atom to afford Pd-hydride (1b) (step 1). This is followed by the adsorption of bicarbonate (HCO3−) generated by the dissociation of CO2 in water (step 2). The resulting reaction intermediate (1c) undergoes hydrogenation by a neighboring hydride species to give another intermediate 1d (step 3). Finally, the production of formate together with H2O (step 4) completes the catalytic cycle22,23,42. In an effort to elucidate the cause of the positive effect of alloying with Ag, potential energy profiles were produced using DFT calculations, employing Pd22 and Pd11Ag11 model clusters (Fig. 7B). In the case of Pd22, the H2 dissociation energy via transition state TS1a/1b was calculated to be 15.8 kcal/mol. After the adsorption of HCO3−, hydrogenation by the neighboring hydride species occurs via TS1c/1d with an energy barrier of 63.1 kcal/mol. The energy barrier in step 4, in which formate is spontaneously produced along with H2O when the OH of the HCO3− is attacked by another Pd-hydride species, is quite low. These results indicate that step 3 is the rate-limiting stage in the present catalytic cycle. In the case of the Pd11Ag11 cluster model, the activation energy for the dissociation of H2 was determined to be 11.9 kcal/mol, while the reduction of HCO3− was found to be occurs with a barrier of 51.2 kcal/mol. These results indicate that alloying with Ag plays an important role in promoting the rate-limiting step 3 rather than the H2 dissociation step 1.
The catalytic activities of Pd$_{1}$Ag$_{2}$-loaded ZIF-8 (Pd$_{1}$Ag$_{2}$@ZIF-8) samples without a core-shell structure and prepared by different deposition methods were also investigated. As shown in Fig. 5, poor catalytic performance was obtained from the Pd$_{1}$Ag$_{2}$@ZIF-8-1 and Pd$_{1}$Ag$_{2}$@ZIF-8-2, due to the relatively large Pd$_{1}$Ag$_{2}$ nanoparticles size, which had mean sizes of 10.5 and 7.6 nm, respectively (Fig. S4). In comparison, the ZIF-8@Pd$_{1}$Ag$_{2}$@ZIF-8 exhibited improved catalytic activity, which can be ascribed to the high degree of dispersion of the Pd$_{1}$Ag$_{2}$ nanoparticles within the ZIF-8 as well as the positive effect of the thin shell protecting the nanoparticles during the reaction process. A TEM image of the ZIF-8@Pd$_{1}$Ag$_{2}$@ZIF-8 after the reaction is shown in Fig. S5. The Pd$_{1}$Ag$_{2}$ nanoparticles evidently remained well dispersed within the ZIF-8, with no significant aggregation, confirming the remarkable stability of this catalyst. To further assess the stability of the ZIF-8@Pd$_{1}$Ag$_{2}$@ZIF-8, the catalyst was recovered from the reaction solution using centrifugation and washed with water. The recycled ZIF-8@Pd$_{1}$Ag$_{2}$@ZIF-8 could be re-used at least three times without a significant loss of activity, as demonstrated in Fig. S6. Based on the above results, it is clear that the present synthetic approach avoids the typical issue of the aggregation of metal nanoparticles on the external surfaces of MOFs as well as prevents damage to the MOF during the post-reduction process. Consequently, the catalytic activity during CO$_{2}$ hydrogenation to produce FA is enhanced.

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**Figure 6.** Representative Mulliken atomic charges and HOMO levels determined by DFT calculations for (A) Pd$_{22}$ and (B) Pd$_{11}$Ag$_{11}$ clusters.

**Figure 7.** (A) A possible reaction pathway and (B) potential energy profiles for CO$_{2}$ hydrogenation to FA over Pd$_{12}$ and Pd$_{11}$Ag$_{11}$ cluster models.
Figure 5 also displays the catalytic activity of the ZIF-8@Pd1Ag2@ZIF-8, as well as those of unsupported Pd1Ag2 nanoparticles and Pd1Ag2-loaded ZIF-8 samples (Pd1Ag2/ZIF-8-1 and Pd1Ag2/ZIF-8-1) prepared by different methods during the dehydrogenation of FA to produce H2. These results are well correlated with the extent of hydrogenation of CO2 to FA over these same materials. No reaction occurred using the unsupported Pd1Ag2 nanoparticles due to the significant aggregation of these nanoparticles in the presence of the FA, as shown in Fig. S7, indicating the importance of the support material to prevent the agglomerations. Again, we can still observe the small amount of colloidal Pd1Ag2 nanoparticle while keeping its original particles size, suggesting that strongly binding PVP ligands prevents the reaction of FA decomposition. Considerable activity was exhibited by the Pd1Ag2@ZIF-8-1, and this activity was enhanced in the case of the ZIF-8@Pd1Ag2@ZIF-8 due to the high dispersion of the Pd1Ag2 nanoparticles within the ZIF-8 and protection of the nanoparticles from agglomeration by the shell.

To verify the positive effect of the alloy in terms of promoting the FA dehydrogenation, potential energy profiles were calculated based on DFT, and are summarized in Fig. 8. The lowest energy FA adsorption structure was the trans-M(O)–M(H-O)-bridged configuration (2a, M = Pd or Ag). The HCOOH adsorbed on the Pd11 cluster model evidently undergoes O–H bond cleavage to produce the formate intermediate 2b via TS2a/2b with an energy barrier of 20.0 kcal/mol (step 1). Subsequently, this species is isomerized to afford a trans-M(H)–Pd(O)-bridged HCOOH configuration structure (2c) via TS2b/2c, with a barrier of 15.2 kcal/mol (step 2). The reaction intermediate 2c then undergoes C–H bond cleavage to form CO2 and a Pd–H species (2d) via TS2c/2d with a barrier of 16.6 kcal/mol (step 3). Following this, the catalytic cycle is completed by H2 release via TS2d/2e with a barrier of 24.6 kcal/mol (step 4). The activation energies for each elementary steps over the Pd11Ag11 cluster model were determined to be 11.9, 13.5, 13.0 and 23.0 kcal/mol for steps 1–4, respectively. Alloying Pd with Ag changes the electron density at the active Pd sites due to charge transfer from Ag to Pd resulting from the different work functions of the two elements. This boosts the intrinsic catalytic performance of Pd for FA dehydrogenation12,20,22,23.

The kinetic isotopic effect was assessed using HCOOH and DCOOH, and the kH/kD value obtained with ZIF-8@Pd1Ag2@ZIF-8 (kH/kD = 2.7) was found to be smaller than that associated with the ZIF-8@Pd3@ZIF-8 (kH/kD = 3.4), suggesting that C–H dissociation is facilitated by the presence of electron-rich Pd arising from the charge transfer from Ag to Pd (Table S3). This role of electron-rich Pd atoms in achieving high catalytic activity has also been reported previously30,63. Interestingly, the kH/kD values were found to be 1.3 and 1.25 for the ZIF-8@Pd1Ag2@ZIF-8 and ZIF-8@Pd3@ZIF-8, respectively. These small values can be ascribed to the positive effect of N atoms within the ZIF-8 framework, which serve as proton scavengers to facilitate the dissociation of the O–H bond in FA to produce a formate intermediate30,41,44,45.

Conclusions

Bimetallic PdAg nanoparticles were successfully encapsulated in ZIF-8 using a “bottle around ship” approach with the aim to prevent the aggregation of the PdAg nanoparticles during the catalytic reaction. For the first time, ZIF-8@PdAg@ZIF-8 was applied to the hydrogenation of CO2 to FA and the dehydrogenation of FA to CO2 and H2. Among investigated, ZIF-8@PdAg@ZIF-8 exhibited the highest catalytic performance for both reactions. Higher or lower Pd:Ag ratios decreased the catalytic activity, demonstrating the optimum relative proportion of
both elements in the nanoparticles. The enhanced catalytic performance of this material can be explained by the synergistic effect of combining the PdAg alloy and ZIF-8 support. The electron-rich Pd sites caused by charge transfer from Ag to Pd as well as the basic N groups within the nanopores of the ZIF-8 play important roles in promoting the formation of FA and H₂. Poor catalytic performance was obtained by the PdₙAgₘ@ZIF-8 due to the formation of relatively large nanoparticles, which confirms the importance of the core-shell structure. Moreover, the catalyst was stable and reusable because the outer shell made of composite prevented the aggregation of the PdAg nanoparticles during the catalytic reaction, such that there was no significant loss of catalytic activity after several recycling trials. The present study may pave the way to the development of practical and environmentally benign reversible hydrogen storage systems based on FA.

**Experimental Section**

**Synthesis of PdₙAgₘ nanoparticles.** PdₙAgₘ nanoparticles (n and m represents the theoretical mole ratio between Pd and Ag) were synthesized by using a reported method with a small modification. In a typical experiment, two solutions were prepared. Taking the Pd₄Ag₂ as an example, solution 1 (containing 0.416 g of polyvinylpyrrolidone (PVP, 30 K) as stabilizing agent, 0.0424 g of AgNO₃ as Ag precursor and 15 mL ethylene glycol as solvent) was stirred at 80 °C for 2 h. Solution 2 (containing 0.02805 g of Palladium (II) acetate as Pd precursor and 6.25 mL of 1,4-Dioxane as solvent) was stirred at room temperature. Solution 1 was cooled to 0 °C and then 0.7 mL of 1 M NaOH solution was added to adjust the pH of the resulting mixture under stirring. Then, solution 2 was poured into solution 1 under vigorous stirring and the final mixture was heated up to 100 °C for 2 h. After the metallic colloid preparation, PdAg nanoparticles were purified by adding an excess of acetone and shaking the solution, which caused the extraction of PVP to the acetone phase and flocculation of the PdAg nanoparticles. The supernatant organic phase was removed and the purified nanoparticles were redispersed in 40 mL of methanol (Pd₄Ag₂: 0.375 mmol). Meanwhile, 0.375 mmol of Pdₙs, PdₙAg₁, and PdₙAg₂ₙₚ nanoparticles dissolved in 40 mL methanol were synthesized with same procedure.

**Synthesis of ZIF-8@PdAg@ZIF-8.** In a typical experimental, 0.037 g of Zn(NO₃)₂ and 0.0103 g of Hmin were respectively dissolved in 10 mL of methanol, the mixture was allowed to react at room temperature for 30 min without stirring, then 0.5 mL (Pd₄Ag₂: 9.4 μmol) of pre-synthesized PdAg nanoparticles solution was added into above mixture, further keep at room temperature without stirring. After 24 h, the product was collected by centrifugation, washed several times with methanol. Finally, the products were dried under vacuum overnight. The obtained sample is noted as ZIF-8@Pd₄Ag₂@ZIF-8. The loaded Pd amount was determined to be 2.98 wt.% by ICP analysis. Meanwhile, ZIF-8@Pd₄Ag₂@ZIF-8 and ZIF-8@Pd₄Ag₈@ZIF-8 were synthesized with same synthetic route. For comparison, the added metal nanoparticles amount was same as ZIF-8@Pd₄Ag₂@ZIF-8.

**Synthesis of Pd₄Ag₂@ZIF-8-1.** Pd₄Ag₂ loaded on the external surface of ZIF-8 (noted as Pd₄Ag₂@ZIF-8-1) were synthesized by introducing the Pd₄Ag₂ with the intention of 2.98 wt.% of Pd in the methanol and ZIF-8 suspension, followed by stirring at room temperature for 6 hours, then the product was collected under vacuum condition at 60 °C.

**Synthesis of Pd₄Ag₈@ZIF-8-2.** Pd₄Ag₈@ZIF-8-2 was synthesized by impregnation method. Typically, 0.2 g of ZIF-8 was dispersed in 20 mL of methanol and sonicated for 15 min. then, a given amount of PdCl₂ (with the intention of 2.98 wt.% of Pd) and AgNO₃ aqueous solution was added and stirred for 6 hours at ambient temperature. After stirring, NaBH₄ (ten times molar ratio of loading metal) was added into above solution and further stirred for 15 min. Finally, the product was collected by under vacuum condition at 60 °C.

**Characterization**

Nitrogen adsorption studies were performed by using BEL-SORP max system (BEL, Japan, Inc.) at 77 K. In order to remove the adsorbed impurities, the samples were degassed in vacuum at 473 K for 24 h prior to analysis. Powder X-ray diffraction (XRD) measurements were conducted by using a RigakuRINT2500 Ultima IV X-ray diffractometer with Cu Kα radiation (λ = 1.5406 Å). TEM micrographs were obtained by using Hitachi HF-2000 field emission-transmission electron microscope (FE-TEM) equipped with Kevex energy-dispersive X-ray detector operated at 200 kV. Pd and Ag K-edge X-ray Absorption Fine Structure (XAFS) were performed using a fluorescence-yield collection technique at the BL01B1 station with an attached Si (111) monochromator at SPring-8, JASRI, Harima, Japan (Prop. No. 2017A1063, 2017A1057). The EXAFS data were normalized by fitting the background absorption coefficient, around the energy region higher than that of the edge of about 35–50 eV, with smooth absorption of an isolated atom. Fourier transformation (FT) of k³-weighted normalized EXAFS data was performed over the range of 3.0 Å < r/Å < 12 Å to obtain the radial structure function. Backscattering amplitude and phase shift parameters for a curve-fitting analysis were theoretically calculated with FEFF8.40 code. DFT calculations were performed with the DMOl³ program in Materials Studio 17.2.97.30 The generalized gradient approximation (GGA) exchange–correlation functional proposed by Perdew, Burke, and Ernzerhof (PBE) was combined with the double-numerical basis set plus polarization functions (DNP). The top layers of Pd₁₁ and Pd₁₁Ag₁₁ cluster models were allowed to relax during geometry optimizations, while the bottom two layers of were fixed at the corresponding bulk position.

**Hydrogenation of CO₂.** The catalytic activity test for hydrogenation of CO₂ to FA was performed in an aqueous solution. Briefly, a sample (50 mg) was suspended in NaHCO₃ aqueous solution (10 mL, 1 M) in an autoclave, and the pressure was increased to 1.0 MPa of CO₂ and then increased to 2.0 MPa with H₂. The reaction system was heated to 373 K and stirred for 24 h. The FA was analysed by HPLC with a Shimazu HPLC instrument equipped with a Bio-radAminergic Analysis Column and an Aminex HPX-87H Ion Exclusion Column. The
mobile phase is H₂SO₄ (5 mM, 0.500 mL/min). The catalytic performance, defined as the mmol of formic acid produced per gram of whole catalyst, was calculated on the basis of the whole catalyst amount and by using the amount of formic acid generated after 24 h.

Dehydrogenation of FA. The catalytic behavior of the ZIF-8@Pd-Ag@ZIF-8 for dehydrogenation of FA was evaluated by using a closed liquid-phase system. 20 mg of each powder samples and 4.8 mL of methanol were mixed into a Pyrex reaction vessel (30 mL) which was sealed with a rubber septum. Afterward, samples were treated in an ultrasound bath for 15 min to disperse the catalysts. Then, the mixture was bubbled with argon gas for 30 min to ensure inert atmosphere. Subsequently, 0.2 mL of FA was added into the vessel with magnetic stirring at oil bath (30°C). The H₂ evolution was monitored by using a Shimadzu GC14B equipped with MS5A column after fixed reaction times. The catalytic performance, defined as the mmol of H₂ produced per gram of whole catalyst, was calculated on the basis of the whole catalyst amount and by using the amount of H₂ generated after 2 h.

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Author contributions
M.W. conceived and performed experiment except DFT calculation. K.M. and H.Y. initiated and supervised the project. K.M. performed DFT calculations. Y.F. helped the sample preparation. Y.K., M.N.-G. and T.A. helped the data analysis. The manuscript was written through the discussion with all authors. All authors have given approval to the final version of manuscript.

Competing interests
The authors declare no competing interests.

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