Nanoparticles capable of mimicking natural tissues represent a major technological advancement in regenerative medicine. In this pilot study, the development of a new nanohybrid composed of titanate nanoribbons to mimic the extracellular matrix is reported. During the first phase, nanoribbons were synthesized by hydrothermal treatment. Subsequently, titanate nanoribbons were functionalized by heterobifunctional polyethylene-glycol (PEG) to graft type I collagen on their surface. Biological properties of this new nanohybrid such as cytotoxicity to cardiac cells and platelet aggregation ability were evaluated. The so-formed nanohybrid permits cellular adhesion and proliferation favoring fine cardiac tissue healing and regeneration.

Introduction

Recently, the market for tissue engineering has reached great heights, with key applications being clinical therapies and tissue modeling. Other applications of tissue engineering are personalized and regenerative medicine, cell-based biosensors etc. A market forecast of surpassing $4.8 billion is being predicted by the year 2028, for a total value of tissue-engineered products with upcoming technologies such as 3D bioprinting and artificial intelligence, decellularization and recellularization of organs and electrospinning.

Regenerative medicine and tissue engineering aim to develop biological substitutes for the reconstruction of damaged tissues and the development of new healthy tissues. Currently, the main agents used for tissue regeneration are composed of proteins from the extracellular matrix (ECM). These molecules arrange themselves in a unique and complex 3D structure having structural and biological properties similar to tissue sources. Polymer networks are also used for tissue reconstruction and regenerative medicine. These structures enable fiber-based self-regeneration of tissue while limiting the risks of rejection and disease transmission strongly present in a transplant tissue from a healthy part of the body. Generally, these networks are made of poly nanofibers [lactic-co-glycolic acid] (PLGA), poly [l-lactic acid] (PLLA), poly(caprolactone) (PCL), poly(ethylene oxide) (PEO) also known as polyethylene glycol (PEG), poly(vinyl alcohol) (PVA), poly(ester urethane) urea (PEUU), gelatin, collagen, protein or fibrinogen.

The use of products based on biological components of human or animal origin, or polymer-based regenerative medicine or tissue reconstruction is not without risk. Those bio-products often cause side reactions, embolisms, allergies, infections, damage to nerves and tissue, transmission of diseases and viruses, the formation of blood clots, toxic and anaphylactic shock, necrosis of certain tissues etc.

Commercial products made of polymers may generate, in turn, the compression of the nerves, tissue damage, inflammation, kidney and neurological damage, cerebrospinal fluids leaks, etc.

Another approach has also been proposed to use inorganic fibers such as carbon nanotubes and nanofibers, TiO₂ nanofibers or nanotubes or gold nanowires for tissue reconstruction and regenerative medicine. However, such structures are cylindrical in shape, with dimensions (length and width) of relatively low values.

A need therefore exists for a new material for use in regenerative medicine and tissue engineering which does not have the drawbacks listed above.

Since the observation of carbon nanotubes by Iijima in 1991, the tubular morphology has been thoroughly studied, especially those derived from titanium oxides. On the other
hand, alternative bidimensional morphologies such as
nanoribbons were rarely studied and were most often considered as
undesired by-products of synthesis or as an intermediate step of
the nanotubes formation. Nevertheless, for about fifteen
years, their specific properties have enabled titanate nano-
ribbons to become a full-fledged recognized nanostructure
more and more studied by scientists.

Titanate materials are of great interest in regenerative
medicine due to their mechanical properties and high resis-
tance to corrosion. Their surfaces are covered with hydroxyl
groups that offers the possibility, compared to polymer-based
products or biological components of human or animal
origin, of functionalizing them with active molecules to couple,
for instance, therapeutic and diagnostic effects for tissue
regeneration. Furthermore, thanks to their radiosensitizing
effect, titanates can also enhance any radiotherapeutic treat-
ment. Elongated shapes of TiO2 nanomaterials such as tita-
nate nanotubes (TiONTs) showed promising healing properties
supported by a promotion of cell growth and proliferation
thanks to TiONTs matrix. Furthermore, in implants, TiONTs
or TiO2 are also providing photocatalytic activities enhancing
their antibacterial properties. These properties can also
reduce low prolonged inflammation reactions once implanted
especially when coated with biopolymer such as collagen.
Collagen has significant applications in tissue engineering.
Owing to its excellent biocompatibility, biodegradability, facile
extraction process, weak antigenicity and purification, scientific
exploration concerning collagen have inspired the field of tissue
engineering.

In this pilot study, TiO2-based materials for regenerative
medicine and tissue engineering were studied. Most particu-
larly titanate nanoribbons (TiONRs) nanostructure with low
cytotoxicity, functionalized with biocompatible active polymers
and structural adhesion proteins (type I collagen) was devel-
oped. In this study, with a focus on cardiac damages, prelimi-
nary aggregation and adhesion measurements on fibroblasts
demonstrated the particular interest of functionalized TiONRs
to promote healing processes and regeneration of damaged
heart’s tissues.

Materials and methods

Chemicals

All chemicals and reagents were of analytical grade and used
without further purification. Methoxy polyethylene glycol 5000 g
mol⁻¹ (mPEG₅₀₀₀ from Sigma Aldrich) and NHS-PEG₅₀₀₀-OH
(JenKem Technology) were silanized to obtain mPEG₅₀₀₀-Si [see
protocol SI_1] and NHS-PEG₅₀₀₀-Si respectively.

Synthesis of TiONRs

The synthesis of TiONRs has been previously described. Briefly, titanate nanoribbons were synthesized by a hydro-
thermal treatment in strongly basic conditions. For this reac-
tion, 110 mL of NaOH aqueous solution at 10 mol L⁻¹ was
prepared and introduced into a sealed Teflon reactor. Then
440 mg of TiO₂ precursor (P25 Degussa) was added to the
solution and the mixture underwent pulsed ultrasound treat-
ment for 30 min at a power of 375 W (Sonics Vibra-Cells). The
hydrothermal treatment took place at 180 °C with an autogenic
pressure (7 bar), for 20 hours and under low mechanical stirring
(150 rpm). The precipitate obtained at the end of the reaction
was separated from the synthesis supernatant by a centrifugation
cycle of 10 min at 11 000 × g. Finally, in order to wash the
powder and to reach a neutral pH, the precipitate was dialyzed
against water (at 3.5 kDa MWCO) for several days before being
freeze-dried prior to characterization.

PEGylation of TiONRs

TiONRs’ surfaces were functionalized with a mixture of
mPEG₅₀₀₀-Si and NHS-PEG₅₀₀₀-Si (see ESI and Fig. SI_1† for
the silanization of PEG derivatives). 10.6 mg of naked TiONRs were
dispersed under manual agitation in dichloromethane. Then
32 mg (mass ratio PEG : TiONRs = 3 : 1) of a molar ratio of 95%
of mPEG₅₀₀₀-Si (30 mg) and 5% of NHS-PEG₅₀₀₀-Si (2 mg) were
added to the TiONRs suspension. The mixture was magnetically
stirred (150 rpm) for 48 h at 20 °C under inert atmosphere. The
excess of PEGs was then washed with centrifugation cycles in
dichloromethane and PEGylated-TiONRs were finally freeze-
dried before further use and characterization. Hereafter, these
nanohybrids are referred to as TiONRs-PEG-NHS.

Functionalization of TiONRs-PEG

Functionalization of TiONRs-PEG-NHS with collagen was per-
formed via NHS ester–amine reaction. Briefly, TiONRs-PEG-
NHS was suspended in PBS at 50 μg mL⁻¹ with type I collagen
(Horn from Nycodem) at 10 μg mL⁻¹ at 20–22 °C for 30
minutes. The suspension was then washed twice by centrifugation
in PBS (4000 × g, 2 min) and the nanohybrids (TiONRs-
PEG-Coll-I) were resuspended in PBS.

Characterizations of TiONRs nanohybrids

Transmission Electron Microscopy (TEM) characterization was
performed using a JEOL JEM-2100F microscope operating at
200 kV (point to point resolution of 0.19 nm). One hundred
nanoribbons were counted in order to calculate nanoribbons
average dimensions.

Powders were analysed using a Discovery TGA-TA Instru-
ments with an air flow rate of 25 mL min⁻¹. A temperature ramp
of 5 °C min⁻¹ from 25 °C to 800 °C was applied.

Specific surface area (SSA) measurements were performed
using a Micromeritics Tristar II apparatus. Samples were out-
gassed in situ at 100 °C under a pressure of 26 mbar for 15 h and
the measurements were performed at liquid N₂ temperature
with N₂ adsorbing gas.

Considering mass losses at different temperatures coupled
to the SSA measurements, grafting rates of the two polymers
were calculated.

Polymer silanization was followed by proton nuclear
magnetic resonance (¹H-NMR). ¹H-NMR spectra of synthesized
polymers were recorded on a Bruker AVANCE 300 spectrometer
in deuterated chloroform (CDCl₃) at 300 MHz and 293 K (see
ESI†).
Cytotoxicity evaluation
With a focus on cardiac damages, MTT cytotoxicity tests were performed on cardiomyocytes and fibroblasts (rat’s primary culture) in contact with naked TiONRs or TiONTs. Dose effect of TiONRs was evaluated for 72 hours. On day 1, the cells in 24-well plates were incubated (37 °C, 5% CO₂) with TiONRs suspensions at 2, 20, and 66 μg mL⁻¹ for 24 h. This operation was repeated twice on day 2 and 3 to study the dose effect (48 h and 72 h). For a morphology comparison, titanate nanotubes (TiONTs: length: 150 nm and diameter: 10 nm) synthesized following protocols from previous studies were also incubated with fibroblasts in the same conditions and at the final concentration of 66 μg mL⁻¹. After incubation, cells were rinsed twice with PBS at 37 °C and incubated for 1 h with 500 μL of MTT at 2 mg mL⁻¹ in PuCK G+ cell medium. Finally, MTT solution was replaced by 500 μL of isopropanol solution with 0.1 mol L⁻¹ HCl for 45 min at 37 °C before optical analysis at 570 nm. The experiments were run in independent triplicate to perform statistical analyses.

Platelet aggregation
Platelet aggregation tests were performed on platelet-rich plasma (PRP) incubated with naked TiONRs. Blood samples were collected from volunteer donors into citrate 3.2% collector tubes (BD Vacutainer France). Tubes were centrifuged at 150 × g for 10 min to obtain PRP. The residual blood was further centrifuged at 2500 × g for 15 min to obtain plasma poor plasma (PPP). The chosen PRP chosen was coming from voluntary donors and was selected to have at least 350 × 10⁶ platelets per μL of plasma. 290 μL of PRP (>300 g L⁻¹) were mixed with 10 μL of naked TiONRs to obtain final concentrations of 1, 10, 25, 50, 100 and 250 μg mL⁻¹. The suspensions of TiONRs in PRP were preincubated for 20 min at 37 °C before the measurements. As a positive control, 10 μL adenosine diphosphate (ADP) at 5 μM was used to stimulate the aggregation of 290 μL of PRP. Aggregation was measured via thrombo-aggregometer (Ta18v from SD Medical). Another experiment was also performed to study the influence of naked TiONRs on platelet aggregation. 290 μL of PRP were mixed with 10 μL of naked TiONRs at final concentrations of 100 and 250 μg mL⁻¹. 11 minutes after reactions in the thrombo-aggregometer, 10 μL of ADP were added to induce the aggregation. Intensities of aggregation were then measured with light transmissions set up at 0% and 100% for PRP and PPP respectively.

Cell adhesion
Effects of TiONRs’ functionalization with collagen on cell adhesion were measured. Type I collagen (Coll-I), naked TiONRs, TiONRs-PEG-NHS, naked TiONRs + 10 μg mL⁻¹ of Coll-I and TiONRs-PEG-Coll-I were placed in wells of 96 wells plates (Maxisorp from Nunc) at 50 μg mL⁻¹ of nanohybrids and 10 μg mL⁻¹ of Coll-I for 2 h at 37 °C. Then, wells were washed twice with PBS and passivated with bovine serum albumin (BSA) at 30 g L⁻¹ in PBS to avoid unspecific binding. The excess of BSA was also washed twice with PBS. MRC-5 cells (human pulmonary fibroblasts) were incubated in the wells with 40 000 cells per well/100 μL at 37 °C for 2 h. Then the wells were washed twice with PBS and the adhesive cells were fixed with 100 μL per well of cold methanol (−20 °C) for 10 min. The cells were coloured with 100 μL per well of crystal violet at 5 mg mL⁻¹ in methanol for 10 min. The reactants were finally washed with water and the cells were dried overnight before dissolving the crystal violet in 100 μL per well of SDS solution (10 mg mL⁻¹) and measuring the absorbance at 570 nm. The values of adhesion were compared to the cell adhesion on 10 μg mL⁻¹ of Coll-I.

Results and discussion
Characteristics of TiONRs
The TiONRs obtained from hydrothermal syntheses have a specific morphology (Fig. 1) with a length between 1 to 20 microns, and a width varying from 70 to 200 nm. These average dimensions were measured on 8 reproducible batches of TiONRs (Fig. SI_2†). The average thickness of TiONRs is comprised between 3 and 40 nm; dimension’s measurements was optimized in a previous publication. The morphology of the TiONRs is particularly suitable to highly and easily cover surface for potential tissue regeneration demonstrated by a good biomimetics with the extracellular matrix. By optimizing parameters, synthesis reaches 99% purity, that means less than 1% (in number) of by-products such as nanosheets, nanotubes or remaining TiO₂ precursor are mixed with nanoribbons. This synthesis is reproducible as there are only few variations in terms of structure, morphology, and chemical composition (type Na₄H₂₋₃Ti₄O₁₀₋₁₁₋ₓH₂O) after close to a dozen of syntheses (see ESI and Fig. SI_2†). Functionalization of TiONRs with Si-PEG-NHS improved the colloidal stability of TiONRs (Fig. 2-a). PEG polymers were chosen because of their high biocompatibility with many other nanomaterials. The success of the silanization by Si-PEG-NHS was confirmed with NMR analyses (Fig. SI_3†) proving a 90% yield of silanization and a final 80% yield of active NHS after functionalization. With a specific surface of 25 m² g⁻¹ for naked TiONRs coupled to the two mass losses (from 100 to 450 °C) of TiONRs-PEG-NHS (17.7% see Fig. SI_4†), corresponding to the degradation of mPEG5000-Si and Si-PEG5000-NHS, the concentrations of these two PEG on TiONRS are 1 and 0.1 molecules per nm², respectively.

Thus, TiONRs have 0.08 active NHS per nm². All the physicochemical characteristics of the TiONRs are summarized in Table 1.

TiONRs as a potential nanohybrid for regenerative medicine
No significant cytotoxicities of TiONRs were found on cardiomyocytes. The maximum toxicity on TiONRs on fibroblasts was below 20% (Fig. 2-b) for the highest concentrations (66 μg mL⁻¹) and no significant toxicity was found for the two other concentrations tested (2 and 20 μg mL⁻¹). This value is more than twice lower than the cytotoxicity observed after incubation of 1 to 3 doses of TiONTs at a similar concentration on both cell lines. This low toxicity of TiONRs could be attributed to their
original morphology. In fact, elongated nanomaterials such as nanotubes or nanorods have usually more chance to be internalized by cells and then to influence cell integrity and cause more damage. Regarding the TiONRs, because their lengths are higher than 1 \( \mu \)m and they are quite larger than classical nanotubes/rods (a few hundreds of nm compared to a few dozens of nm) they have less chance to be internalized and to promote cytotoxicity compared to the nanotube morphology that showed significantly higher cell killing effect.\(^{47,48}\) Furthermore, a dose effect was observed with TiONTs when none was observed after 3 successive doses of TiONRs.

To confirm the hypothesis of lower cellular interactions of TiONRs compared to TiONTs, cytotoxicity measurements were correlated with optical microscopy pictures of fibroblasts (Fig. 3-a) incubated with nanoparticles. Two similar doses of TiONRs at 66 \( \mu \)g mL\(^{-1}\) showed much more aggregates of particles on top of the cells than with 2 doses of TiONTs at 66 \( \mu \)g mL\(^{-1}\) confirming that cells might not have internalized many TiONRs compared to TiONTs.

For concentrations up to 250 \( \mu \)g mL\(^{-1}\), TiONRs did not induce any spontaneous aggregation of the platelets in the PRP compared to the ADP which induced a rapid increase of aggregation within 3 min (Fig. 2-c). At a relatively high concentration (<100 \( \mu \)g mL\(^{-1}\)), TiONRs do not have any side effect on platelet aggregation which allows their use as matrix for regenerative medicine. However, at 250 \( \mu \)g mL\(^{-1}\), TiONRs seem to slow down the platelet activity, as evidenced by the 2-fold reduction of the measured aggregation compared to the control and the TiONRs at 100 \( \mu \)g mL\(^{-1}\). A concentration of 250 \( \mu \)g mL\(^{-1}\) of TiONRs seems to be too high and might affect platelet aggregation.

Cellular adhesion efficiency of fibroblasts in the presence of TiONRs with or without PEG and functionalized or not with Coll-I was then quantified (Fig. 3-b). Surface covered with Type I collagen was used as a positive control. Coll-I is the most abundant collagen in human body made of fibers used to heal

### Table 1  TiONRs physicochemical characterizations

| TiONRs-PEG-NHS | Length (\( \mu \)m) | Width (nm) | Thickness (nm) | Purity (%) | SSA (m\(^2\) g\(^{-1}\)) | mPEG-Si (nm\(^{-2}\)) | NHS-PEG-Si (nm\(^{-2}\)) | NHS (nm\(^{-2}\)) |
|----------------|-----------------|------------|----------------|------------|----------------|----------------|----------------|----------------|
| 1–20           | 70–200          | 3–40       | 99             | 25         | 1              | 0.1            | 0.08           |                |

Fig. 1  TEM pictures of titanate nanoribbons.

Fig. 2  (a) Photograph and TEM pictures showing the dispersion of the naked and PEG-functionalized TiONRs; (b) cytotoxic dose effect of naked TiONRs and TiONTs at highest concentration on fibroblasts and cardiomyocytes. * Significant differences compared to the control \( p < 0.05 \); $ significant differences compared to the TiONRs \( p < 0.05 \); (c) aggregation’s activation of platelets in presence of TiONRs at 1, 10, 25, 50, 100 and 250 \( \mu \)g mL\(^{-1}\) and ADP at 5 \( \mu \)M. ADP was injected 11 minutes after incubation. Platelets aggregation’s activation in presence of TiONRs at 100 and 250 \( \mu \)g mL\(^{-1}\) was also measured. ADP: adenosine diphosphate.
wounds and already well used to improve cellular adhesion in regenerative medicine. Among the non-toxic concentrations, a TiONRs concentration of 50 μg mL⁻¹ was chosen to cover the whole plate, thus limiting modulation of the platelet aggregation and forming a monolayer (the optimization of which was corroborated by optical microscopy: see Fig. SI_5†).

Surface coated with naked and PEGylated TiONRs do not allow cellular adhesion. 27% of adhesion were observed on naked TiONRs mixed with Coll-I. As TiONRs are completely covering the plate, this adhesion could certainly be explained by non-specific adsorption of the collagen on TiONRs surface allowing some cells to stick on the plate surface. Such unspecific adsorption is difficult to control and very common with proteins and nanomaterials. For the chemically functionalized TiONRs-PEG-Coll-I, the percentage of adhesion is as good as the adhesion with only Coll-I alone, proving the efficacy of the grafting of collagen on PEGylated TiONRs. As PEG is a well-known polymer that prevents proteins adsorption and that is also used for its antifouling properties preventing cellular adhesion, the obtained rate of adhesion proved that PEG is not accessible on the surface of the plate. It also proved successful covalent functionalization of Coll-I on TiONRs-PEG-NHS that allowed successful adhesion of fibroblasts on well-plate. Besides, it can be noted that the Coll-I coating prevents the direct interaction of the TiONRs’ scaffold with fibroblasts.

Conclusions
In this study, a new nanobiohybrid favoring cellular adhesion and proliferation has been developed for fine tissue healing and regeneration, especially on cardiac damages. In a pilot study, we were able to synthesize in a reproducible manner titanate nanoribbons (TiONRs) with biocompatible functional polymers and type I collagen. TiONRs showed low cytotoxicity against cardiomycocytes and cardiac fibroblasts with a minimal dose effect compared to nanotube morphology. At concentrations below 100 μg mL⁻¹, TiONRs neither induced nor inhibit the platelet adhesion opening the way for their putative use as bandages to cover skin wounds. We demonstrated that well controlled functionalization of TiONRs could lead to future materials for regenerative medicine with potential theranostics’ applications thanks to the interesting properties of titanate material.

Author contributions
Study design: VB, DV and NM. TiONRs synthesis, functionalization and characterization: VB, MM, JB, NM and FD. Biological studies: VB, LD, EDM and DV. LM and VB analysed the data and prepared figures from the experiments. Writing – Original Draft Preparation: LM; Writing – Review & Editing: LM, VB, NM, JB, FD, LD, EDM and DV; Funding Acquisition: NM and DV.

Conflicts of interest
The authors declare no conflict of interest.

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Notes and references

1. P. X. Ma, *Adv. Drug Deliv. Rev.*, 2008, 60, 184–198.
2. S. Ullah and X. Chen, *Appl. Mater. Today*, 2020, 20, 100656.
3. H. N. Abdelhamid and A. P. Mathew, *Int. J. Mol. Sci.*, 2022, 23, 5405.
4. M. A. Nilforoushzadeh, M. A. Amirkhani, E. Khodaverdi, Z. Razzaghi, H. Afzali, S. Izadpanah and S. Zare, *Tissue Cell*, 2022, 74, 101717.
5. R. Mohammadinejad, A. Kumar, M. Ranjarb-Mohammadi, M. Ashrafizadeh, S. S. Han, G. Khang and Z. Roverimia, *Polymers*, 2020, 12, 176.
6. J. Y. Ng, S. Obuobi, M. L. Chua, C. Zhang, S. Hong, Y. Kumar, R. Gokhale and P. L. R. Ee, *Carbohydr. Polym.*, 2020, 241, 116345.
7. A. X. Y. Guo, L. Cheng, S. Zhan, W. Xiong, Z. Wang, G. Wang and S. C. Cao, *J. Mater. Sci. Technol.*, 2022, 125, 252–264.
8. N. Tsao, presented in part at the Opportunities for living tissue equivalents and technologies for their manufacture, November, 2017.
9. F. Guilak, D. M. Cohen, B. T. Estes, J. M. Gimble, W. Liedtke and C. S. Chen, *Cell Stem Cell*, 2009, 5, 17–26.
10. G. S. Hussey, J. L. Dziki and S. F. Badylok, *Nat. Rev. Mater.*, 2018, 3, 159–173.
11. Z. Pan and J. Ding, *Interface Focus*, 2012, 2, 366–377.
12. E.-M. Engelhardt, L. A. Micl, S. Houis, F. M. Wurm, J. Hilborn, J. A. Hubbell and P. Frey, *Biomaterials*, 2011, 32, 3969–3976.
13. E. A. Phelps, N. Landazuri, P. M. Thule, W. R. Taylor and A. J. Garcia, *Proc. Natl. Acad. Sci. U. S. A.*, 2010, 107, 3323–3328.
14. R. Hartwell, V. Leung, C. Chavez-Munoz, L. Nabai, H. Yang, F. Ko and A. Ghahary, *Acta Biomater.*, 2011, 7, 3060–3069.
15. J. M. Anderson, A. Rodriguez and D. T. Chang, *Semin. Immunol.*, 2008, 20, 86–100.
16. A. L. Predeina, M. S. Dukhninova and V. V. Vinogradov, *J. Mater. Chem. B*, 2020, 8, 10010–10022.
17. M. Griffith, M. M. Islam, J. Edin, G. Papavenglou, O. Buznyk and H. K. Patra, *Front. Bioeng. Biotechnol.*, 2016, 4, 71.
18. M. M. J. P. E. Stijnis, M. J. Jetten, S. G. Mohammed, S. M. H. Claessen, R. H. W. de Vries, A. Stell, D. F. A. de Bont, M. A. Engelse, D. Muncucoglu, C. A. van Blitterswijk, P. Y. W. Dankers, E. J. P. de Koning, A. A. van Apeldoorn and V. L. S. LaPointe, *Biomaterials*, 2021, 267, 120449.
19. P. A. Tran, L. Zhang and T. J. Webster, *Adv. Drug Deliv. Rev.*, 2009, 61, 1097–1114.
20. S. Qian, Z. Yan, Y. Xu, H. Tan, Y. Chen, Z. Ling and X. Niu, *RSC Adv.*, 2019, 9, 12001–12009.
21. X. Wang, R. A. Gittens, R. Song, R. Tannenbaum, R. Olivares-Navarrete, Z. Schwartz, H. Chen and B. D. Boyan, *Acta Biomater.*, 2012, 8, 878–885.
22. K. Cowden, M. F. Dias-Netipanyj and K. C. Popat, *Nanomed. Nanotechnol. Biol. Med.*, 2019, 17, 380–390.
23. D. Fan, Z. Yin, R. Cheong, F. Q. Zhu, R. C. Cammarata, C. L. Chen and A. Levchenko, *Nat. Nanotechnol.*, 2010, 5, 545–551.
24. A. M. Brokesh and A. K. Gaharwar, *ACS Appl. Mater. Interfaces*, 2020, 12, 5319–5344.
25. S. K. Nethi, S. Das, C. R. Patra and S. Mukherjee, *Biomater. Sci.*, 2019, 7, 2652–2674.
26. S. Iijima, *Nature*, 1991, 354, 56–58.
27. Inorganic and Metallic Nanotubular Materials: Recent Technologies and Applications, ed. T. Kijima, Springer-Verlag, Berlin Heidelberg, 2010.
28. Y. Zhang, Z. Jiang, J. Huang, L. Y. Lim, W. Li, J. Deng, D. Gong, Y. Tang, Y. Lai and Z. Chen, *RSC Adv.*, 2015, 5, 79479–79510.
29. D. V. Baykin, J. M. Friedrich and F. C. Walsh, *Adv. Mater.*, 2006, 18, 2807–2824.
30. F. Sallem, R. Chassagnon, A. Megriche, M. El Maouoi and N. Millot, *J. Alloys Compd.*, 2017, 722, 785–796.
31. A.-L. Papa, N. Millot, L. Saviot, R. Chassagnon and O. Heintz, *J. Phys. Chem. C*, 2009, 113, 12682–12689.
32. Y. Dong, Z.-S. Wu, S. Zheng, X. Wang, J. Qin, S. Wang, X. Shi and X. Bao, *ACS Nano*, 2017, 11, 4792–4800.
33. K. Kordás, M. Mohl, Z. Kónya and Á. Kukovecz, *Transl. Mater. Res.*, 2015, 2, 015003.
34. A. Loiseau, J. Boudon, A. Oudot, M. Moreau, R. Boidot, R. Chassagnon, N. M. Said, S. Roux, C. Mirjolet and N. Millot, *Cancers*, 2019, 11, 1962.
35. C. Mirjolet, A. L. Papa, G. Créhange, O. Raguin, C. Seignez, C. Paul, G. Truc, P. Maingon and N. Millot, *Radiother. Oncol.*, 2013, 108, 136–142.
36. N. A. Ismail, K. A. Mat Amin and M. H. Razali, *Mater. Lett.*, 2018, 228, 116–120.
37. X. Zheng, J. Sun, W. Li, B. Dong, Y. Song, W. Xu, Y. Zhou and L. Wang, *J. Electroanal. Chem.*, 2020, 871, 114362.
38. A. Khalid, H. Ullah, M. Ul-Islam, R. Khan, S. Khan, F. Ahmad, T. Khan and F. Wahid, *RSC Adv.*, 2017, 7, 47662–47668.
39. Y. Zhao, L. Bai, Y. Zhang, R. Yao, Y. Sun, R. Hang, X. Chen, H. Wang, X. Yao, Y. Xiao and R. Hang, *Biomaterials*, 2022, 121684.
40. R. Parenteau-Bareil, R. Gauvin and F. Berthod, *Materials*, 2010, 3, 1863–1887.
41. V. Bellat, R. Chassagnon, O. Heintz, L. Saviot, D. Vandrour and N. Millot, *Dalton Trans.*, 2015, 44, 1150–1160.
42. A.-L. Papa, J. Boudon, V. Bellat, A. Loiseau, H. Bisht, F. Sallem, R. Chassagnon, V. Bérard and N. Millot, *Dalton Trans.*, 2015, 44, 739–746.
43. A.-L. Papa, L. Dumont, D. Vandrour and N. Millot, *Nanotoxicology*, 2012, 7, 1131.
44. M. T. Wolf, K. A. Daly, J. E. Reing and S. F. Badylok, *Biomaterials*, 2012, 33, 2916–2925.
45. A. P. Khandhar, P. Keselman, S. J. Kemp, R. M. Ferguson, P. W. Goodwill, S. M. Conolly and K. M. Krishnan, *Nanoscale*, 2017, 9, 1299–1306.
46 C. Sun, K. Du, C. Fang, N. Bhattarai, O. Veiseh, F. Kievit, Z. Stephen, D. Lee, R. G. Ellenbogen, B. Ratner and M. Zhang, **ACS Nano**, 2010, **4**, 2402–2410.

47 P. Foroozandeh and A. A. Aziz, **Nanoscale Res. Lett.**, 2018, **13**, 339.

48 R. Augustine, A. Hasan, R. Primavera, R. J. Wilson, A. S. Thakor and B. D. Kevadiya, **Mater. Today Commun.**, 2020, **25**, 101692.

49 K. R. Levental, H. Yu, L. Kass, J. N. Lakins, M. Egeblad, J. T. Erler, S. F. T. Fong, K. Csiszar, A. Giaccia, W. Weninger, M. Yamauchi, D. L. Gasser and V. M. Weaver, **Cell**, 2009, **139**, 891–906.

50 G. Chen, Y. Yu, X. Wu, G. Wang, J. Ren and Y. Zhao, **Adv. Funct. Mater.**, 2018, **28**, 1801386.

51 M. Mahmoudi, I. Lynch, M. R. Ejtehadi, M. P. Monopoli, F. B. Bombelli and S. Laurent, **Chem. Rev.**, 2011, **111**, 5610–5637.

52 M. J. Hajipour, S. Laurent, A. Aghaie, F. Rezaee and M. Mahmoudi, **Biomater. Sci.**, 2014, **2**, 1210–1221.

53 Y. Li, R. Lin, L. Wang, J. Huang, H. Wu, G. Cheng, Z. Zhou, T. MacDonald, L. Yang and H. Mao, **J. Mater. Chem. B**, 2015, **3**, 3591–3603.

54 R. Gref, M. Lück, P. Quellec, M. Marchand, E. Dellacherie, S. Harnisch, T. Blunk and R. H. Müller, **Colloids Surf. B Biointerfaces**, 2000, **18**, 301–313.

55 I. Banerjee, R. C. Pangule and R. S. Kane, **Adv. Mater.**, 2011, **23**, 690–718.

56 S. Rigo, C. Cai, G. Gunkel-Grable, L. Maurizi, X. Zhang, J. Xu and C. G. Palivan, **Adv. Sci.**, 2018, **5**, 1700892.