Effects of intercalated atoms on electronic structure of graphene nanoribbon/hexagonal boron nitride stacked layer

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Using first-principles calculations, we investigate an atomic impurity at the interface of a van der Waals heterostructure (vdW heterostructure) consisting of a zigzag graphene nanoribbon (ZGNR) and a hexagonal boron nitride (h-BN) sheet. To find effects of atomic intercalation on geometrical and electronic properties of the ZGNR on the h-BN sheet, various types of impurity atoms are considered. The embedded atoms are initially placed at the edge or the middle of the ZGNR located on the h-BN sheet. Our results demonstrate that most of the impurity atoms are more stable at the edge than at the middle in all cases we consider. Especially, a nickel atom has the smallest energy difference (~0.15 eV) between the two embedding positions, which means that the Ni atom is relatively easy to intercalate in the structure. Finally, we discuss magnetic properties for the vdW heterostructure with an intercalated atom.

Graphene has attracted tremendous attention as a next-generation electronic material because of its unique chemical and physical properties. Mica, silicon carbide (SiC), and silicon dioxide (SiO₂) have been usually used as substrates for nanodevices. However, such types of substrates reduce the electronic mobility of graphene in the device. Recently, the concern with the hexagonal boron nitride (h-BN) sheet as an insulating substrate for the graphene devices has been growing. When graphene is transferred directly to the conventional substrates such as SiO₂, the electrical mobility is decreased by charged impurities in the substrates. Besides, the surface roughness of the SiO₂ substrate results in the large corrugation of graphene. In contrast, graphene on the h-BN sheet is almost flat, and shows much higher electrical mobility than graphene on SiO₂. The h-BN sheet is an insulator with a large optical bandgap of about 6 eV and small lattice mismatch with graphene about 1.6%. Another important point is that h-BN has weak van der Waals (vdW) interaction with graphene. Therefore, h-BN may be a good candidate as a dielectric substrate for graphene-based nanodevices. Besides, this material is chemically inert, compared to metal surfaces.

The free-standing zigzag graphene nanoribbon (ZGNR) shows localized electronic states and antiferromagnetic property at the edge. The edge states of the ZGNR can be affected by impurity atoms or the supporting substrate. Using first-principles calculations, Choi et al. studied the adsorption properties of alkali metal atoms from the edge to the middle of ZGNRs. The adsorption energies of alkali metal atoms on the ZGNR depend on the position of the impurity, and are largest at the ZGNR edge. On the other hand, Lee et al. considered the functionalization of halogen atoms and molecules at the ZGNR edge. They focused on free-standing ZGNR for the adsorption of atomic impurities. Since the real nanodevices are fabricated on a substrate, however, we need to study intercalation properties of such defect atoms at interfaces of graphene/substrate systems.

In this paper, we report a first-principles study of atomic and electronic properties of a ZGNR on an h-BN sheet with an intercalated atom. The purpose of our work is to understand intercalation properties of extrinsic defects such as alkali metal (Li, Na, and K) and halogen (Cl, Br, and I) atoms in order to tailor the localized states of the zigzag edge in a graphene on an h-BN sheet. We also examine effects of other types of residual atoms (Cu, Ni, and Si atoms) between graphene and h-BN sheet, which could be adsorbed to the graphene or h-BN sheet in...
the transfer process. The intercalated atoms are expected to bring about charge doping or to change the magnetic properties of the system.

**Results and Discussion**

**Van der Waals ZGNR/h-BN heterostructures.** First, we obtain the optimized geometry of the vdW ZGNR/h-BN heterostructure with no intercalated atom, as shown in Fig. 1(a), where the grey, blue, and pink colours represent carbon, nitrogen, and boron atoms, respectively. The most stable configuration resembles the Bernal stacking of graphite. The carbon atoms of the ZGNR are located on top of the boron atoms and the centre of h-BN hexagons; that is, the centres of the ZGNR hexagons are on top of the nitrogen atoms of h-BN. Our results are in consistent with the previous reports\(^2\)\(^{20-26}\). The equilibrium distance between the ZGNR and the h-BN sheet is 3.15 Å. We find that the localized states exist at the edges of the ZGNR on the h-BN sheet, represented by the almost flat band of the zigzag edge around the Fermi level (E\(_F\)). The localized states\(^2\)\(^7\), however, rapidly decay into the bulk, as shown in Fig. 1(b). Figure 1(b) shows the projected densities of states (PDOSs) at the left and right edges and near the middle of the ZGNR, respectively. The red, green, and yellow lines of the PDOSs of the ZGNR correspond to the same colours of carbon atoms at the ZGNR in Fig. 1(a), respectively. The
results are same as those for free-standing ZGNRs\(^\text{28}\). The PDOS plots imply that the h-BN sheet does not affect the electronic properties of the ZGNR, in contrast to the conventional substrates such as SiC and SiO\(_2\). We also calculate spin densities to examine the magnetic properties of the edge states of the ZGNR. As in the ZGNR without a substrate, antiferromagnetic properties are clearly shown in the ZGNR on the h-BN sheet, which means that spin-up and spin-down electrons are localized at each edge of the ZGNR, as shown in Fig. 1(c).

For the ZGNR placed on the h-BN sheet, we assume that the ZGNR edges are terminated by hydrogen atoms. The space between the H-terminated edge of the ZGNR and the h-BN sheet may open rather wide, and the open-edged ZGNR on h-BN may allow atoms or molecules to be intercalated between the ZGNR and the h-BN sheet. We place the alkali (Li, Na, and K), halogen (Cl, Br, and I), and other impurity (Cu, Ni, and Si) atoms at both edge and deep positions. We find the stable positions for the intercalated atoms located at edge and deep positions. The binding energy of each impurity atom between the ZGNR and the h-BN sheet is defined as

\[
E_b = E_{\text{tot}}(\text{impurity}) + E_{\text{tot}}(\text{ZGNR/h-BN}) - E_{\text{tot}}(\text{ZGNR/impurity/h-BN}),
\]

where \(E_{\text{tot}}\) represents the total energy of the optimized geometry for each system. Here, the positive (negative) sign of the binding energy represents an exothermic (endothermic) process. The binding energy values are listed in Table 1. Our results show that the deep position is less stable than the edge position because of the large curvature and strain resulted from the deformation of the graphene and h-BN sheet. For all the halogen atoms we consider, the deep position has negative binding energies, and is energetically unstable. The Li atom is small, and may be incorporated at the deep position since the difference of the binding energy is only ~0.21 eV between the deep and edge positions. We find the stable positions for the intercalated atoms located at edge and deep positions.

| Alkali | Halogen | Others |
|--------|---------|--------|
| Edge   | Deep    | Edge   | Deep   | Edge   | Deep   |
| Li     | 2.61    | Cl     | 2.09   | Li     | 1.10   |
| Na     | 1.46    | Br     | 1.63   | Na     | 1.59   |
| K      | 1.24    | I      | 1.18   | K      | 2.09   |

Table 1. Binding energy (in eV) of the intercalated atoms between the ZGNR and the h-BN sheet.

\(\text{K}\) and \(\text{Br}\) atoms for a detailed study. Figure 2 shows top and side views of optimized geometries and the PDOSs of the ZGNR/impurity/h-BN structures for K and Br impurities. Obviously, the interstitial space between the ZGNR and the h-BN sheet is enlarged in the presence of the intercalated atom at both edge and deep positions, as shown in Fig. 2. As shown in Fig. 2(a,b), neither h-BN nor ZGNR makes a chemical bond with the K atom for both edge and deep positions. The red, yellow, and green lines of the PDOS correspond to the left edge, right edge, and deep positions of ZGNRs, respectively, and the black line of the PDOS represents the impurity atom. Electron transfer occurs from the K atom to the ZGNR and the h-BN sheet at both edge and deep positions. At the edge position, the K atom has weak orbital hybridization with the ZGNR around \(E_p\), as shown in Fig. 2(a). Here, we find the broad potassium bands in the conduction band (~1.8 eV). It means that the K atom is fully ionized, and donates an electron to the ZGNR. Interestingly, the left edge near the K impurity does not have any practical spin-polarized state, whereas the right edge still has a spin-polarized state like a defect-free ZGNR edge. For both edge and deep positions, the strong edge state of the ZGNR occurs at \(E_p\) and the trend of electron donation is expected to be very similar. Therefore, one could use the partially filled edge states as one-dimensional channels for quantum electronic transport.

In contrast to the potassium impurity, the bromine atom forms a chemical bond to a carbon atom of the left edge of the ZGNR with sp\(^3\)-like hybridization, and the left edge states of the ZGNR disappear in the edge-position configuration. Because of strong electron affinity of Br, the ZGNR is hole-doped, and the PDOS of the Br atom is clearly shown between ~3.0 and ~2.0 eV. More interestingly, the Br atom at the deep position creates spin magnetic moments coupled to the ZGNR edge, which will be discussed below in more detail, although it forms no chemical bonds to the ZGNR or the h-BN sheet. At the deep position, the narrow DOS peaks of Br occur near \(E_p\) for down-spins, and the localized edge states of the ZGNR are shown near \(E_p\) in Fig. 2(d). Thus, the Br-originated localized state near \(E_p\) can cause electron scattering.

Van der Waals ZGNR/h-BN heterostructures with alkali metal and halogen atoms. To investigate the electronic structures of the intercalated atoms at the interface of the ZGNR/h-BN structure, we calculate the atomic and electronic structures of alkali metal and halogen atoms placed at both edge and deep positions, as shown in Fig. 2. Similar trends in atomic and electronic structures are shown for alkali metal and halogen atoms. We choose the K and Br atoms for a detailed study. Figure 2 shows top and side views of optimized geometries and the PDOSs of the ZGNR/impurity/h-BN structures for K and Br impurities. Obviously, the interstitial space between the ZGNR and the h-BN sheet is enlarged in the presence of the intercalated atom at both edge and deep positions, as shown in Fig. 2. As shown in Fig. 2(a,b), neither h-BN nor ZGNR makes a chemical bond with the K atom for both edge and deep positions. The red, yellow, and green lines of the PDOS correspond to the left edge, right edge, and deep positions of ZGNRs, respectively, and the black line of the PDOS represents the impurity atom. Electron transfer occurs from the K atom to the ZGNR and the h-BN sheet at both edge and deep positions. At the edge position, the K atom has weak orbital hybridization with the ZGNR around \(E_p\), as shown in Fig. 2(a). Here, we find the broad potassium bands in the conduction band (~1.8 eV). It means that the K atom is fully ionized, and donates an electron to the ZGNR. Interestingly, the left edge near the K impurity does not have any practical spin-polarized state, whereas the right edge still has a spin-polarized state like a defect-free ZGNR edge. For both edge and deep positions, the strong edge state of the ZGNR occurs at \(E_p\) and the trend of electron donation is expected to be very similar. Therefore, one could use the partially filled edge states as one-dimensional channels for quantum electronic transport.

In contrast to the potassium impurity, the bromine atom forms a chemical bond to a carbon atom of the left edge of the ZGNR with sp\(^3\)-like hybridization, and the left edge states of the ZGNR disappear in the edge-position configuration. Because of strong electron affinity of Br, the ZGNR is hole-doped, and the PDOS of the Br atom is clearly shown between ~3.0 and ~2.0 eV. More interestingly, the Br atom at the deep position creates spin magnetic moments coupled to the ZGNR edge, which will be discussed below in more detail, although it forms no chemical bonds to the ZGNR or the h-BN sheet. At the deep position, the narrow DOS peaks of Br occur near \(E_p\) for down-spins, and the localized edge states of the ZGNR are shown near \(E_p\) in Fig. 2(d). Thus, the Br-originated localized state near \(E_p\) can cause electron scattering.
configuration for Cu, Ni and Si defects in order to check the impurity effect on the electronic properties and spin magnetic properties of the ZGNR, although the edge configuration is energetically more stable than the deep-position configuration. Figure 3 shows the atomic and electronic structures of the vdW ZGNR/h-BN heterostructures with Cu, Ni, and Si defects at edge and deep positions, respectively. The Si atom results in significant deformation of the ZGNR and the h-BN sheet, but the Cu and Ni atoms do not cause remarkable geometrical changes in the two layers at the deep positions. The large and narrow PDOSs originating from Cu 3d orbitals are shown around $-3$ eV, and the Cu 4 s electron is slightly hybridized with the ZGNR edge states at $E_F$. For both the edge and deep positions, electron transfer takes place from the Cu atom to the ZGNR, and cancels the spin magnetic moment of the left edge of the ZGNR, as shown in Fig. 3(a,b). For the Ni impurity, the Ni 3d orbitals are hybridized with the localized C 2p orbitals at the ZGNR edge between $-2.0$ and $0$ eV ($E_F$), as shown in Fig. 3(c,d). Figure 3(e) shows that the Si impurity atom bonds to the ZGNR edge and makes DOS peak just below $E_F$. On the other hand, the Si atom at the deep position produces two nearly-degenerate localized states just below $E_F$, as shown in Fig. 3(f).

Magnetic properties of van der Waals ZGNR/h-BN heterostructures with impurity atoms. Finally, we explore spin magnetic properties of the ZGNR/impurity/h-BN structures. Figure 4 shows the spin densities and the PDOS for ZGNR/h-BN structures with intercalated atoms (K, Br, Cu, Ni and Si) at both edge and deep positions. The red and yellow colours represent the spin-up and spin-down electrons, respectively. Table 2 summarizes the spin magnetic moment of each system. ZGNRs have localized states at the edge, and have the opposite spin configuration between two edges\(^{19}\). For alkali metal atoms at the edge position, the net spin magnetic moment of the system is $\sim 1.0 \mu_B$ (the Bohr magneton, $1 \mu_B \approx 9.274 \times 10^{-24}$ J T$^{-1}$). For the edge position, it is interesting that electron transfer takes place from the incorporated alkali metal atom to a ZGNR edge so that the spin magnetic moment of the zigzag edge bonded to the alkali metal atom disappears; however, the spin configuration of the opposite edge retains, as shown in Fig. 4(a). For the deep position, the antiferromagnetic

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**Figure 2.** Atomic and electronic structures of vdW ZGNR/h-BN heterostructures with intercalated (a,b) potassium and (c,d) bromine atoms at the edge and deep positions, respectively. Each colour line shown at the ZGNRs correspond to the same colour solid line of the PDOS. The red, green, and yellow atoms at the ZGNR represent the 1st, 5th, and 18th line from the left ZGNR edge, respectively.
configuration in the ZGNR is almost kept [see Fig. 4(b)], but electrons donated from the alkali metal atom are redistributed to give magnetic moment of 0.2 μB.

For the edge position, on the other hand, the halogen atom forms a chemical bond to an edge of the ZGNR, and the spin magnetic moment results only from the opposite site of the ZGNR, which gives the spin magnetic moment of 1.0 μB. Compared with Fig. 1(b), Fig. 4(c) shows that the localized states at the left edge (A and B states in the Fig. 1(b)) disappear near E_F. Very interestingly, although the ZGNR keeps an antiferromagnetic configuration for the deep position, an almost vertical distribution of spin-up electrons around the halogen atom occurs as shown in Fig. 4(d), which gives the spin magnetic moment of ~1.0 μB. For the Cu impurity atom, the net spin magnetic moments at both the edge and deep positions are about 1.0 μB. As shown in Fig. 4(e), for the edge position, the left edge of the ZGNR near the Cu atom cancels the spin magnetic moment, and the remaining

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**Figure 3.** Atomic and electronic structure of the ZGNRs on the h-BN sheet with intercalated (a, b) Cu, (c, d) Ni, and (e, f) Si atoms at the edge and deep positions, respectively. The carbon atoms of red, green, and yellow colours at the ZGNRs correspond to the same colour solid lines of PDOS. The red, green, and yellow atoms at the ZGNRs represent the 1st, 5th, and 18th line from the left ZGNR edge, respectively.
Figure 4. Spin densities and the PDOS for (a,b) K, (c,d) Br, (e,f) Cu, (g,h) Ni, and (i,j) Si atoms at the edge and deep positions, respectively. The red and yellow balloons represent the spin-up and spin-down states, respectively. The black, red, and blue dotted lines in the PDOS represent the ZGNR, impurity, and h-BN sheet, respectively.
spin configuration of the right edge gives a magnetic moment. The PDOS also reveals that a filled spin-up state [state A in Fig. 1(b)] moves upward, while an unfilled spin-down state [state B in Fig. 1(b)] moves downward. They meet at the same energy so that the spin density at the left edge disappears. An interesting point is that the Cu atom at the deep position also affects the spin magnetic moment of the left ZGNR edge, as shown in Fig. 4(f), although the distance between the Cu atom and the left edge is about 1.0 nm. We thus find a similar trend to the case of the edge position that a spin-down edge state moves downward and a spin-up state moves upward in energy for the left edge. The Ni defect induces no net magnetic moment, as previously reported in literature30,31, and the ZGNR retains antiferromagnetic, as shown in Fig. 4(g,h). However, it is evidently found that strong hybridization between the ZGNR and h-BN sheets occurs owing to the Ni 3d electrons. On the other hand, the left edge states of the ZGNR almost disappear when the Si atom is bonded to the left edge of ZGNR. Consequently, the net spin magnetic moment becomes zero. The PDOS in Fig. 4(i) shows strongly hybridized states between the Si atom and ZGNR just below $E_F$ for spin-up electrons. The Si atom at the deep position induces the spin magnetic moment of $\sim 1.6 \mu_B$, while the ZGNR maintain the antiferromagnetic configuration. We find the difference between the PDOS of the edge and deep positions. For the deep position, the states A and B in Fig. 1(b) re-appear in Fig. 4(i), compared to the edge position, and the spin-up states of the Si atom around $E_F$ are hybridized with the h-BN states as well as the ZGNR states. We find the difference between the PDOSs of the edge and deep positions. For the deep position, the edge states of the ZGNR move downward, and two almost-degenerate Si states appear just below $E_F$. Therefore, the net spin magnetic moment of $\sim 1.6 \mu_B$ comes mainly from the Si atom, as shown in Figs 3(i) and 4(j).

Conclusions

We have carried out the first-principles calculations to investigate the electronic and atomic structure of the ZGNRs on the h-BN sheet with intercalated atoms such as alkali, halogen, and other atoms. The localized states of a ZGNR edge on the insulating h-BN sheet were observed to show a rapid decay into the bulk. Intercalated atoms located at edge and deep positions are considered. The deep position is less stable than the edge position because of strong deformation of the ZGNR and h-BN sheet. We also investigate the magnetic properties of the systems. In particular, the spin magnetic moments come from the ZGNR edge for the alkali metal defect, whereas they come from the intercalated atoms for the halogen impurity. The incorporating atoms affect the ZGNR edge in the vdW ZGNR/h-BN heterostructure. Therefore, it is important to identify the modification of the edge states of the ZGNR on the h-BN sheet with intercalated atoms in comparison with the free-standing ZGNR and to remove the residual atoms on ZGNRs for the devices.

Methods

First-principles calculations. To understand the interaction between the ZGNR and the h-BN sheets with an intercalated atom, we have performed density functional theory (DFT) calculations within generalized gradient approximation (GGA) using the Vienna ab initio simulation package (VASP)32–34. The projector augmented wave potentials are employed to describe the potential from the centre of each atom. The energy cut-off for the plane-wave potentials are optimized until the Hellman-Feynman forces acting on the atoms become smaller than 0.03 eV/Å. To include weak vdW interaction between the adsorbate and the ZGNR, we adopt Grimme’s DFT-D2 vdW corrections based on a semi-empirical GGA-type theory35. For the Brillouin-zone interaction, we use a $\Gamma$-centered $1 \times 3 \times 1$ grid in the Monkhorst-Pack k-point scheme.

For the study of the doping effect of intercalated atoms, we choose alkali metal (Li, Na, and K) and halogen (Cl, Br, and I) atoms. Alkali metal atoms tend to donate an electron to graphene, whereas halogen atoms are likely to accept one electron from graphene36–38. In addition, we consider Cu, Ni, and Si atoms to investigate the changes caused by the atoms in atomic and electronic structures of the vdW ZGNR/h-BN heterostructure since Cu, Ni, and SiC are often used for growing graphene. If the atoms originating from the substrate surface are not removed perfectly during transfer, the graphene or h-BN sheet may contain the impurity atoms on its surface. To study effects of residual atoms, therefore, we consider the impurities such as Cu, Ni, and Si atoms.

The supercell size is $65.33 \times 10.06 \times 25 \AA^3$. The vdW ZGNR/h-BN heterostructure consists of a stacked structure of a single layered h-BN sheet and the $N = 18$ ZGNR. The width of the ZGNR is about 38 Å. Two types of positions of the impurity atom are considered. One is just below an edge of the ZGNR that is labeled the edge position, and the other is $\sim 1.0$ nm far inside the zigzag edge that is labeled the deep position.

References

1. Novoselov, K. S. et al. Electric field effect in atomically thin carbon films. *Science* **306**, 666–669 (2004).
2. Morozov, S. et al. Giant intrinsic carrier mobilities in graphene and its bilayer. *Phys. Rev. Lett.* **100**, 016602 (2008).
3. Castro Neto, A. et al. The electronic properties of graphene. *Rev. Mod. Phys.* **81**, 109 (2009).

| Alkali | Halogen | Others |
|-------|---------|--------|
| Edge  | Deep    | Edge  | Deep   | Edge  | Deep  |
| Li    | 0.94    | Cl    | 1.00   | Cu    | 0.98  |
| Na    | 0.98    | Br    | 1.00   | Ni    | 0.00  |
| K     | 0.98    | I     | 1.00   | Si    | 1.63  |

Table 2. Net spin magnetic moments (in $\mu_B$) of the ZGNR/impurity/h-BN structures at edge and deep positions.
4. Zomer, P. et al. A transfer technique for high mobility graphene devices on commercially available hexagonal boron nitride. Appl. Phys. Lett. 99, 232104 (2011).
5. Chun, H. et al. Ultraflat Graphene. Nature 462, 339–341 (2009).
6. Berger, C. et al. Electronic confinement and coherence in patterned epitaxial graphene. Science 312, 1191–1196 (2006).
7. Brar, V. W. et al. Scanning tunneling spectroscopy of inhomogeneous electronic structure in monolayer and bilayer graphene on SiC. Appl. Phys. Lett. 91, 122102 (2007).
8. Zhang, Y., Tan, Y.-W., Stormer, H. L. & Kim, P. Experimental observation of the quantum Hall effect and Berry’s phase in graphene. Nature 438, 201 (2005).
9. Ishigami, M., Chen, J., Cullen, W., Fuhrer, M. & Williams, E. Atomic structure of graphene on SiO2. Nano Lett. 7, 1643–1648 (2007).
10. Zhang, Y. et al. Giant phonon-induced conductance in scanning tunnelling spectroscopy of gate-tunable graphene. Nat. Phys. 4, 627–630 (2008).
11. Deshpande, A., Bao, W., Miao, F., Lau, C. N. & LeRoy, B. J. Spatially resolved spectroscopy of monolayer graphene on SiO2. Phys. Rev. B 79, 205411 (2009).
12. Martin, J. et al. Observation of electron-hole puddles in graphene using a scanning single-electron transistor. Nat. Phys. 4, 144–148 (2008).
13. Zhang, Y., Brar, V. W., Girir, C., Zettl, A. & Crommie, M. F. Origin of spatial charge inhomogeneity in graphene. Nat. Phys. 5, 722–726 (2009).
14. Hwang, E., Adam, S. & Sarma, S. D. Carrier transport in two-dimensional graphene layers. Phys. Rev. Lett. 98, 186806 (2007).
15. Katsnelson, M. & Geim, A. Electron transport on microscopic corrugations in graphene. Philosophical Transactions of the Royal Society of London A: Mathematical, Physical and Engineering Sciences 366, 195–204 (2008).
16. Jung, S. et al. Evolution of microscopic localization in graphene in a magnetic field from scattering resonances to quantum dots. Nat. Phys. 7, 245–251 (2011).
17. Watanabe, K., Taniguchi, T. & Kanda, H. Direct-bandgap properties and evidence for ultraviolet lasing of hexagonal boron nitride single crystal. Nat. Mater. 3, 404 (2004).
18. Son, Y.-W., Cohen, M. L. & Louie, S. G. Energy gaps in graphene nanoribbons. Phys. Rev. Lett. 97, 216803 (2006).
19. Son, Y.-W., Cohen, M. L. & Louie, S. G. Half-metallic graphene nanoribbons. Nature 446, 342–347 (2007).
20. Elwing, B. et al. Energy Bandgap and Edge States in an Epitaxially Grown Graphene/B–BN Heterostructure. Sci. Rep. 6, 31160 (2016).
21. Choi, S.-M. & Jhi, S.-H. Self-assembled metal atom chains on graphene nanoribbons. Phys. Rev. Lett. 101, 266105 (2008).
22. Lee, H., Cohen, M. L. & Louie, S. G. Selective functionalization of halogens on zigzag graphene nanoribbons: A route to the separation of zigzag graphene nanoribbons. Appl. Phys. Lett. 97, 233101 (2010).
23. Sachs, B., Wehling, T., Katsnelson, M. & Lichtenstein, A. Adhesion and electronic structure of graphene on hexagonal boron nitride substrates. Phys. Rev. B 84, 195414 (2011).
24. Park, S., Park, C. & Kim, G. Interlayer coupling enhancement in graphene/hexagonal boron nitride heterostructures by intercalated defects or vacancies. J. Chem. Phys. 140, 134706 (2014).
25. Kang, S.-H., Kim, G. & Kwon, Y.-K. Is hexagonal boron nitride always good as a substrate for carbon nanotube-based devices? Phys. Chem. Chem. Phys. 17, 5072–5077 (2015).
26. Lee, J., Kim, M., Chelikowsky, J. R. & Kim, G. Computational simulation of subatomic-resolution AFM and STM images for graphene/hexagonal boron nitride heterostructures with intercalated defects. Phys. Rev. B 94, 035447 (2016).
27. Park, C., Ihm, I. & Kim, G. Decay behavior of localized states at reconstructed armchair graphene edges. Phys. Rev. B 88, 045403 (2013).
28. Fujita, M., Watanabashiki, K., Nakada, K. & Kusakabe, K. Peculiar localised state at zigzag graphite edge. J. Phys. Soc. Jpn. 65, 1920–1923 (1996).
29. Lopina, G. et al. Residual metallic contamination of transferred chemical vapor deposited graphene. ACS nano 9, 4776–4785 (2015).
30. Delkoy, Y. S. & Fonin, M. Electronic and magnetic properties of the graphene–ferromagnet interface. New J. Phys. 12, 125004 (2010).
31. Sung, D., Park, N., Kim, G. & Hong, S. Enhanced binding strength between metal nanoclusters and carbon nanotubes with an atomic nickel defect. Nanotechnology 23, 205204 (2012).
32. Kresse, G. & Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. Phys. Rev. B 54, 11169 (1996).
33. Kresse, G. & Furthmüller, J. Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. Comp. Mater. Sci. 6, 15–50 (1996).
34. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. Phys. Rev. Lett. 77, 3865 (1996).
35. Grimme, S. Semiempirical GGA-type density functional constructed with a long-range dispersion correction. J. Comp. Chem. 27, 1787–1799 (2006).
36. Baede, A., Moutinho, A., De Vries, A. & Los, J. Total cross sections for charge transfer between Alkali Atoms and Halogen Molecules. Chem. Phys. Lett. 3, 530–533 (1969).
37. Avci, R. & Flynn, C. Halogen charge-transfer excitations in alkali metals. Phys. Rev. B 19, 5981 (1979).
38. Davydov, S. Y. & Sabirova, G. Adsorption of hydrogen, alkali metal, and halogen atoms on graphene: Adatom charge calculation. Tech. Phys. Lett. 37, 515–518 (2011).

Acknowledgements
This research was supported by Priority Research Center Program (2010-0020207) through the National Research Foundation of Korea (NRF) funded by the Ministry of Education and Basic Science Research Program (2017R1A2B2010123) and Global Research and Development Center Program (2018R1A4A3A01064272) through NRF funded by the Ministry of Science and ICT. This work was also supported from NRF [2013R1A1A2090131 (G.K.), 2017R1D1A1B03033707 (D.S.)].

Author Contributions
D. Sung did first-principles calculations for the model systems. D. Sung, G. Kim, and S. Hong interpreted the calculated results, and wrote the paper. All authors read and approved the final manuscript.

Additional Information
Supplementary information accompanies this paper at https://doi.org/10.1038/s41598-019-39719-9.

Competing Interests: The authors declare no competing interests.

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