Fast polarization mechanisms in the uniaxial tungsten-bronze relaxor strontium barium niobate SBN-81

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The high-frequency dielectric response of the uniaxial strontium barium niobate crystals with 81% of Sr has been studied from 1 kHz to 30 THz along the polar c axis by means of several techniques (far infrared, time domain terahertz, high-frequency and low-frequency dielectric spectroscopies) in a wide temperature interval 20–600 K. Relaxor properties were observed in the complex dielectric response and four main excitations were ascertained below the phonon frequencies. These fast polarization mechanisms take place at THz, GHz and MHz ranges and show different temperature evolution. The central mode excitation in the THz range, related to anharmonic dynamics of cations, slightly softens from high temperatures and then hardens below \( T \approx 400 \) K. Below the phase transition (at \( T \approx 330 \) K) an additional microwave excitation appears near 10 GHz related to micro domain wall oscillations. The strongest relaxation appears in the GHz range and slows down on cooling according to the Arrhenius law. Finally, another relaxation, present in the MHz range at high temperatures, also slows down on cooling at least to the kHz range. These two relaxations are due to polar fluctuations and nanodomains dynamics. Altogether, the four excitations explain the dielectric permittivity maximum in the kHz range.

The relaxor compositions of the solid solution strontium barium niobate \( \text{Sr}_x\text{Ba}_{1-x}\text{Nb}_2\text{O}_6 \) (SBN) display compelling dielectric, pyroelectric, electro-optic and non-linear optical properties1,2. But, in addition, they present a great challenge in the understanding of the relaxor properties, in comparison to the better studied cubic perovskite relaxors3. SBN crystallizes in the unfilled tetragonal tungsten-bronze (TTB) structure in the compositional range \( 0.32 < x < 0.82 \)4. The TTB structure is a network of interconnected \( \text{NbO}_6 \) octahedra forming three types of channels along the polar c axis which have different site symmetries: the \( A_1 \) site (inside the squared channels) is occupied by Sr, the \( A_2 \) site (inside the pentagonal ones) by Sr and Ba, and the \( C \) site (inside the triangular channels) is empty4,6. The intrinsic disorder present in the structure is the key to its excellent piezoelectric and dielectric properties1,7. The structure is highly anisotropic, as well as all its properties, and compositions with more than \( \sim 50\% \) Sr show already mixed ferroelectric-relaxor properties. The composition with \( \sim 80\% \) of Sr, SBN-80, displays a pronounced relaxor behaviour6: the dielectric permittivity displays a broad maximum in \( \varepsilon'(T) \) at the temperature \( T_m \approx 300 \) K in the sub-hertz range which shifts up to 350 K at several megahertz, and presumably to higher temperatures at higher frequencies. This strong dielectric dispersion, observed mainly at temperatures around the \( \varepsilon'(T) \) maximum, is the main characteristic of the relaxor behaviour6,9. At very high temperatures SBN-80 should show the centrosymmetric point group \( 4/mmm \)10, as other TTBs, although its structure was not actually measured at such high temperatures. Nevertheless, modulations in the position of oxygen atoms are present up to at least 700 K12. On cooling and below \( T_m \), a tetragonal phase appears, with the space group \( P4bm \)6 and spontaneous polarization along the c axis, as in other members of the family with lower Sr content. However, depending on the crystallization method an orthorhombic distortion could appear in some crystals with high Sr content5,13. The size of the ferroelectric domains is dependent on the Sr content and, for \( \text{Sr}_{0.75}\text{Ba}_{0.25}\text{Nb}_2\text{O}_6 \) (SBN-75), the domain size is of the order of 100 nm14. Recently it was found that polar nanoregions in SBN-80 show

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a dynamics in the low-frequency range, namely their transverse and longitudinal breathing, and their transformation into polar nanodomains (PNDs). However, no dielectric data were published at higher frequencies, and therefore faster polarization mechanisms were not included in the picture of the dielectric response of high-Sr content SBN. We already reported the far infrared (IR) reflectivity spectrum of Sr$_{0.61}$Ba$_{0.39}$Nb$_2$O$_6$ (SBN-61) and the temperature dependence of its IR active modes along the polar axis, as well as its broadband dielectric behaviour (from 100 Hz to 67 THz). In this composition, the slowing down of a strong relaxation, together with a THz central mode, is responsible for the huge dielectric anomaly below phonon frequencies.

The fast dynamics of the phase transition in the SBN structure was studied by Raman spectroscopy for crystals with several compositions. A proper soft mode was never found; therefore, an order-disorder mechanism of the transition was implied. In agreement with this, central peaks at very high temperatures were discovered by Brillouin and neutron scattering, indicating the thermal switching of PNDs and the anharmonic hopping motion of Ba/Sr atoms. Inelastic neutron scattering on SBN-61 confirmed the relaxation of polar fluctuations in the $ab$ plane, with a correlation length of 5–10 unit cells, responsible for the dielectric anomaly. These data show that a complex interaction between slow and fast polarization mechanisms is responsible for the phase transition in TTBs.

The dielectric behaviour of the relaxor compositions of SBN should be very dependent on the formation of polar regions of different sizes seen by light scattering and transmission electron microscopy, but up to now no comprehensive and exhaustive dielectric studies were performed in the relaxor compositions. Therefore, in this paper, we report on the dielectric response from $10^3$ to $10^{13}$ Hz along the polar $c$ axis of the utmost uniaxial SBN relaxor with 81% of Sr (SBN-81), in the temperature range 20 to 600 K, using different spectroscopic techniques. This broadband approach is able to unveil the main excitations responsible for the fast mechanisms of the polarization, and elucidate their relationship to the relaxor behaviour.

**Experimental Results**

**IR and THz measurements.** The unit cell of SBN, presented in Fig. 1, has five (Sr$_x$Ba$_{1-x}$)$_5$Nb$_{10}$O$_{30}$ units, and contains 45 atoms, resulting in the formula (Sr$_x$Ba$_{1-x}$)$_5$Nb$_{10}$O$_{30}$ due the vacancies present in the channels. In the paraelectric phase, the theoretical analysis for the centrosymmetric space group $P4/mmm$ predicts 138 phonon modes at the centre of the Brillouin zone:

\[
\Gamma^{\text{TET}}(4/mmm): 9A_{2u}(c; -) + 10A_{2g}(-; -) + 5A_{1u}(-; -) + 10A_{1g}(-; a^2 + b^2, c^2) + 8B_{1u}(-; -) + 10B_{1g}(-; a^2 - b^2) + 4B_{2u}(-; -) + 10B_{2g}(-; ab) + 26E_a(a, b; -) + 10E_g(-; ac, bc).
\]

Out of them, $1A_{2g} + 1E_a$ are acoustic modes, and $E_a$ and $E_g$ modes are double degenerated. IR and Raman activities are shown in parentheses. The site group analysis for the ferroelectric phase, using Jamieson data for SBN-75 at room temperature – point group $4mm^2$, with an effective Ba/Sr atom in the pentagonal site $4c$ and without disorder among O atoms, gives the following active modes:

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**Figure 1.** Structure of SBN in the paraelectric phase in the $ab$ plane and IR reflectivity spectra of SBN-81 (dotted) together with their fits (lines) at several temperatures for the $E||c$ polarization, including the experimental TDTTS data in the THz range. Red spectra are taken in the paraelectric phase, blue in the ferroelectric one.
All modes are optical, except 1 A1 + 1 E, which are acoustic, E modes being double degenerated. However, taking into account oxygen disorder, we obtain 168 active modes distributed into the following phonon symmetries:

\[\Gamma_{\text{TET}}(4mm)_{\text{ordered}}: 19 A_1(c; a^2 + b^2, c^2) + 14 B_1(-; a^2 - b^2) + 15 A_2(-; -) + 18 B_2(-; ab) + 36 E(a, b; ac, bc)\]

\[\Gamma_{\text{TET}}(4mm)_{\text{disordered}}: 23 A_1(c; a^2 + b^2, c^2) + 18 B_2(-; a^2 - b^2) + 19 A_2(-; -) + 22 A_1(-; ab) + 43 E(a, b; ac, bc).\]

External phonon modes due to Ba and Sr vibrations are present in the IR and Raman spectra below \(\nu \sim 180\ \text{cm}^{-1}\). At higher wavenumbers, phonons are due to Nb-O stretching and O-Nb-O bending vibrations in the octahedral network. In the case that SBN had an orthorhombic distortion, the expected point group is mm2 and the 138 phonon modes would be distributed in the following symmetries:

\[\Gamma_{\text{ORT}}(mm 2)_{\text{ordered}}: 33 A_1(c; a^2, b^2, c^2) + 33 A_2(c, ab) + 36 B_1(a; ac) + 36B_1(b; bc).\]

Out of them, 1A1 + 1B1 + 1B2 would be acoustic.

The IR reflectivity spectra of vibrations of SBN-81 along the polar axis, \(E|\epsilon|E\), corresponding to modes of \(A_{2u}/A_1\) symmetry, are depicted in the right panel of Fig. 1. Spectra including the THz reflectivity points calculated from the time-domain THz transmission spectroscopy (TDTTS) experiment– are shown together with the corresponding fits at selected temperatures. The experimental reflectivity spectra were fitted with the generalized oscillator model of the dielectric function:

\[\varepsilon_i(\nu) = \varepsilon'(\nu) - i\varepsilon''(\nu) = \varepsilon_\infty \prod_{i=1}^{n} \frac{v^2_{i\text{T}0\text{K}} - \nu^2}{v^2_{i\text{T}0\text{K}} - \nu^2 + \mathrm{i} \gamma_{i\text{T}0\text{K}}} \]

\[\Delta \varepsilon_i = \frac{\varepsilon_\infty \prod_{i=1}^{n} (v^2_{i\text{T}0\text{K}} - v^2_{i\text{T}0\text{K}})}{v^2_{i\text{T}0\text{K}} - v^2_{i\text{T}0\text{K}}} \]

where \(\varepsilon_\infty\) is the permittivity at frequencies much higher than all phonon frequencies, \(\nu_{i\text{T}0\text{K}}\) and \(\gamma_{i\text{T}0\text{K}}\) are the transverse and longitudinal frequencies of the \(i\)-th phonon mode and \(\gamma_{i\text{T}0\text{K}}\) and \(\gamma_{i\text{T}0\text{K}}\) their respective damping constants. The dielectric strength of each mode \(\Delta \varepsilon_i\) is then automatically calculated by eq. (1b).

In the paraelectric phase, 11 A2u modes were fitted (3 external modes from cations plus 7 internal ones and one overdamped low-frequency mode in the THz range). Theoretically 8 A2u(c) optical modes are allowed, after subtracting the acoustic branch, but the double occupancy of some sites and disorder in oxygen positions enables the activation of more modes. In the ferroelectric phase, 20 A1 modes were fitted at the lowest temperature (5 external modes, 14 internal modes and a low-frequency one in the THz range); although 22 A1 are predicted for the tetragonal phase, and 33 in the case of the orthorhombic variation. According to the reflectivity results, our SBN-81 crystal is macroscopically tetragonal, as there is no evident splitting of phonons even at the lowest temperatures, like in orthorhombic Ba3NaNb5O15. The overdamped THz central mode shows a pronounced softening towards \(T_m\) therefore, it is partially responsible for the phase transition.

Table 1 presents the list of the fitted phonon parameters – transverse frequency \(\nu_{i\text{T}}\), transverse damping \(\gamma_{i\text{T}}\) and dielectric strength \(\Delta \varepsilon_i\), calculated from (1b), in the paraelectric (400 K) and in the ferroelectric (12 K) phases. \(\Delta \varepsilon_i\) is the lowest-frequency mode, in the THz range, changes from 570 at 400 K to 12 at 12 K, and its frequency rises from 20 to 47 cm\(^{-1}\). In the ferroelectric phase new modes, stemming from the formerly Raman active modes in the paraelectric phase, appear. The number of modes and the spectral shape are in agreement with the tetragonal symmetry, as in SBN-61.

The permittivity and losses calculated from the previous fits are presented together with the experimental TDTTS points in Fig. 2, at selected temperatures. The permittivity along the polar axis due to the phonon contribution is rather low (\(\Delta \varepsilon_{\text{phonon}} \sim 40\)), as in other TTB materials, but the THz central mode enhances dramatically the \(\varepsilon'\) values near 10 cm\(^{-1}\) at high temperatures, as seen in Fig. 2a. On cooling, this excitation loses its dielectric strength and the THz values of the permittivity agree with the phonon contribution. In Fig. 2b the dielectric loss spectra are displayed at different temperatures. Each peak corresponds to a phonon mode. On cooling, phonons become better distinguished and the intense THz central mode, seen as a broad peak at high temperatures, gradually weakens and shifts to higher frequency. The oscillator model of eq. (1) used for this fitting, combining reflectivity, permittivity and dielectric loss in the THz range, lacks the influence of the lower frequency relaxations on the GHz range and below, therefore another fit was performed (see Discussion), taking into account lower frequency data, down to kHz. Nevertheless, the model of eq. (1) gives more reliable phonon parameters with reasonable values in the THz range.

### Low-and high-frequency dielectric measurements

The complex dielectric function \(\varepsilon_i(T, \nu) = \varepsilon' - i\varepsilon''\) at high-frequencies obtained by the coaxial technique on cooling is shown in Fig. 3. Upper panels show the strong dispersion of the permittivity and dielectric loss maxima at \(T_m\), shifting from 400 K at 1.8 GHz to 330 K at 1 MHz, and...
it reveals the strong relaxor behaviour of the crystal along the polar axis. The complete set of measured data, with all the frequencies is displayed in the lower panels in the form of 2D intensity maps, where a broad maximum crossing the GHz–MHz ranges is well visible with maximum of intensity around 330 K at megahertzs. The dielectric loss

| SBN-81 E∥c A₁/A₂u modes | Ferroelectric phase (12 K) | Paraelectric phase (400 K) |
|---------------------------|---------------------------|---------------------------|
| νₜ | τₜ | Δε | νₜ | τₜ | Δε |
| νₜme |
| 46.9 | 32.0 | 12 | 19.8 | 52.4 | 57.0 |
| External modes |
| 61.3 | 26.7 | 14.0 | 75.0 | 66.5 | 2.7 |
| 80.3 | 20.9 | 4.0 | 108.9 | 37.0 | 10.0 |
| 133.4 | 48.6 | 7.0 | 148.6 | 78.9 | 10.7 |
| 166.8 | 28.9 | 0.1 | |
| 204.6 | 32.3 | 0.6 | 193.7 | 64.2 | 1.7 |
| 237.0 | 39.1 | 3.8 | |
| 264.5 | 30.1 | 9.6 | 256.0 | 80.9 | 5.0 |
| 284.5 | 21.1 | 1.0 | 291.0 | 100.0 | 3.5 |
| 309.2 | 26.3 | 0.02 | |
| 327.9 | 27.4 | 0.5 | |
| 351.9 | 24.0 | 0.6 | 360.5 | 97.0 | 0.6 |
| 358.5 | 65.0 | 1.0 | |
| 444.0 | 20.4 | 0.1 | |
| 542.1 | 43.8 | 0.7 | 557.0 | 91.2 | 2.0 |
| 622.1 | 58.3 | 2.7 | 637.0 | 95.0 | 1.4 |
| 734.0 | 44.1 | 0.03 | 735.0 | 52.0 | 0.01 |
| 775.0 | 46.0 | 0.01 | |
| 904.0 | 63.0 | 0.02 | |

Table 1. Parameters of the IR phonons of SBN-81 for the E∥c polarization in the paraelectric and ferroelectric phases (νₜ and τₜ in cm⁻¹).

Figure 2. Permittivity (a) and dielectric loss (b) functions of SBN-81 obtained from the far IR fit of the reflectivity at different temperatures, together with the points measured in the TDTTS experiment.
spectrum versus frequency is shown in the lower right panel, where a slowing down of the main relaxation is revealed.

The dielectric measurements up to $10^6$ Hz show that there is an important pre-history effect in the dielectric response of the SBN-81 crystal (see Fig. 4). On heating, apart from the main dielectric anomaly near 330 K, we

**Figure 3.** Upper panels: Temperature dependence of the permittivity (left) and dielectric loss (right) of SBN-81 measured at high frequencies on cooling. Lower panels: 2D intensity maps of the dielectric data –permittivity (left), dielectric loss (right) –in log scale versus temperature and frequency. Colors are illustrative, dark red meaning the highest value.

**Figure 4.** Dielectric measurements at low frequencies of SBN-81 on heating and cooling.
found other anomalies, visible below 1 kHz, at about 280 and 480 K (Fig. 4a). When heating the sample to 600 K and then cooling again (Fig. 4b), these extra anomalies are not seen, just the main anomaly at $T_m$ is present. It seems that the heating erases the thermal and cluster pre-history and transforms the sample into a virgin-like state. After the heating up to 600 K and during the subsequent cooling, the sample showed much higher conductivity. Figure 4c shows the thermal hysteresis of the permittivity at 100 Hz for two consecutive heating-cooling runs. Due to these effects on the values of $\varepsilon'$, the interpretation of the data at very low frequencies are left out of the scope of this paper.

To better understand the frequency dependence of the complex permittivity, we plot together dielectric spectra obtained by the dielectric and TDTTS experiments for selected temperatures. The broad-band experimental dielectric spectra are presented in Fig. 5 with the fits described in the Discussion section. On cooling from 470 K, the low-frequency permittivity rises towards $T_m$ and then decreases monotonically. Maxima of the dielectric loss spectra allow us to identify the main excitations in the crystal. Three distinct peaks are seen at high temperatures in the paraelectric phase. The peak located at the THz range –labelled $\nu_{THz}$– corresponds to the central mode measured in the THz-far IR spectra. The maxima of the other two relaxations –labelled $\nu_0$ and $\nu_2$, respectively– shift to lower frequencies on cooling. The lowest frequency one, $\nu_2$, shifts below the MHz range at about 350 K. Due to the limited amount of data from the dielectric loss spectra at these frequencies, $\nu_2$ is difficult to assess and its parameters are taken only as estimations. On further cooling, below $T_m$, a new excitation ($\nu_{DW}$) appears near 10 GHz. This excitation is needed to reach the permittivity values at lower frequencies and to fit the increase of the dielectric loss data signalized by the slope near the GHz range in Fig. 5b.

### Discussion

To evaluate the behaviour of all the excitations on cooling, we fit the dielectric spectra below phonons at several temperatures with a model consisting of a sum of several Cole-Cole terms for the excitations below THz, a damped oscillator for the central mode in the THz range, and the phonon contribution:

$$\varepsilon'(\nu) = \varepsilon'(\nu) - i\varepsilon''(\nu) = \sum_j \frac{\Delta\varepsilon_{0j}}{1 + (\nu/\nu_0)^{1-\alpha_j}} + \frac{\Delta\varepsilon_{THz}}{\nu_{THz}^2 - \nu^2 + i\gamma_{THz}} + \Delta\varepsilon_{ph}$$

Here, $\Delta\varepsilon_{0j}$ denotes the dielectric strength of the $j$-th excitation, $\nu_0$ its frequency, $\alpha_j$ is a real index between 0 and 1, which determines the deviation from the pure Debye model for the relaxations and gives a notion of the frequency width of the loss peak. In the oscillator term, $\gamma_{THz}$ refers to its damping constant, and $\Delta\varepsilon_{ph}$ stands for the overall contribution of harmonic phonons at higher frequencies, which is about 40.

In Fig. 6a we show the temperature dependence of the polar phonon frequencies detected by the far IR reflectivity experiment and corresponding to the fits using eq. (1). The mean relaxation frequencies found by TDTTS and the dielectric spectroscopies, corresponding to the fits with eq. (2), are shown in Fig. 6b. The phase transition is well detected by phonons between 300 and 325 K. New phonons are activated in the non-centrosymmetric phase. In the THz range, below 50 cm$^{-1}$, there is an overdamped mode which shows an anomaly near 400 K. This mode was fitted with the renormalized law.

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**Figure 5.** Selected frequency cuts of the permittivity and dielectric losses of SBN-81 from the dielectric and coaxial line experiments, together with the experimental points from the TDTTS experiment, and fits of the data using several relaxations. Note the logarithmic scales.
below 400 K; this dependence does not correspond to the classic soft mode behaviour, which evidences that this vibration is not related to the harmonic vibration of an atom, but rather to the anharmonic hopping of disordered cations (Sr, Ba)21. The fitting parameters of this excitation are $T_m \approx 409 \text{ K}$ and $a = 3.4 \times 10^9 \text{s}^{-1} \text{K}^{-1}$. Above $T_m$ there are not enough points to perform a reliable fit, but from Fig. 2b it is evident that the frequency of this excitation hardens on heating from $T_m$. In analogy to SBN-61 and other relaxor ferroelectrics, it can be expected that, on heating, $\nu_{THz}$ increases, merges with phonons in the form of a central mode and, finally, disappears above the Burns temperature, where the PNDs vanish (above 580 K in the case of SBN)9,15.

In Fig. 6b we show the excitations present below phonons. Apart from the THz central mode, there are several excitations at lower frequencies. Because of experimental limitations, we can follow them properly only below 480 K, although they should be present also above this temperature. At high temperatures, in the paraelectric phase, there are probably two excitations: the THz central mode $\nu_{THz}$ and the MHz-GHz range relaxation $\nu_{01}$, which contributes to the permittivity values measured in the dielectric experiment, and is related to dynamics of the polar fluctuations6. On cooling both excitations should slow down and, below 500 K, a third excitation, $\nu_{02}$ is also detected. This relaxation slows down to 100 kHz range at 350 K and corresponds to the relaxation $f_c$ found in8, assigned to the headways breathing of PNDs along the polar axis. We see it at higher temperatures and frequencies, but on cooling it goes out from our frequency window slowing down further into the kHz range. On further cooling, below about 330 K, the fourth excitation $\nu_{DW}$ appears near 10 GHz.

The relaxations $\nu_{01}$ and $\nu_{02}$ slow down from the paraelectric phase, but the excitations $\nu_{THz}$ and $\nu_{DW}$ display different temperature dependences. $\nu_{THz}$ softens towards ~400 K and then hardens, as described above; while $\nu_{DW}$ appears below $T_m$ and exhibits only a slight temperature dependence. $\nu_{DW}$ correlates with the presence of diffuse scattering, as found in SBN-7024 and in SBN-6123. As it is needed to explain the permittivity values only below $T_m$, it is probably due to oscillations of the tiny ferroelectric domains appearing in the crystal, as demonstrated by the partial vanishing of the diffuse scattering by poling23 and by recent studies of the dielectric response of other disordered ferroelectrics29.

Usually, the relaxor behaviour of a material is verified by the analysis of the temperature dependences of the mean relaxation frequency and checking whether it follows the classic Arrhenius law

$$\nu(T) = \nu_{\infty} \exp(-E_a/kT),$$

or the Vogel-Fulcher law, with a freezing temperature $T_{VF}$:

$$\nu(T) = \nu_{\infty} \exp(-E_a/k(T - T_{VF})).$$

where $\nu_{\infty}$ is the saturation frequency, $E_a$ the activation energy and $k$ the Boltzmann constant $k = 8.617 \times 10^{-5}$ eVK$^{-1}$. Figure 7 shows the $\nu(T)$ dependence for the main relaxation, $\nu_{01}$, obtained from the positions of the dielectric loss maxima in both $\varepsilon''(\nu)$ and $\varepsilon''(T)$ shown in Figs 3 and 5b respectively. Although the $\varepsilon''(\nu)$ and $\varepsilon''(T)$ dependences are obtained from the same set of experimental data, the $\nu(T)$ shape is dependent on the selection
of $\varepsilon''(\nu)$ or $\varepsilon''(T)$ maxima, and it could be misleading. In our case, the $\nu(T)$ plot based on the $\varepsilon''(\nu)$ maxima follows the Arrhenius law, while the plot based on the $\varepsilon''(T)$ maxima follows the Vogel-Fulcher law. The Arrhenius fit parameters are $\nu_\infty = 10 \pm 3$ THz and $E_a/\kappa = 3.7 \pm 0.1$ K, whereas the Vogel-Fulcher fit parameters of $\varepsilon''(T)$ are $\nu_\infty = (20 \pm 5)$ GHz, $E_a/\kappa = 236 \pm 30$ K and $T_{VF} = 303 \pm 4$ K. Such a discrepancy is usual for relaxor ferroelectrics (see30, for instance) and reflects the fact that the dielectric strength of the main relaxation in relaxors is decreasing on cooling. Therefore, the correct way to estimate $\nu(T)$ is using the $\varepsilon''(\nu)$ spectra measured at a constant temperature. Consequently, the Arrhenius law for the $\nu_{01}$ relaxation is reliable: the PNDs responsible for the relaxation $\nu_{01}$ in SBN-81 actually do not freeze and the activation energy of their fluctuations is much lower than in SBN-61.

For comparison, the $\varepsilon''(\nu)$ maxima in SBN-61 15 are also displayed in Fig. 7. The fit with eq. (5) [$\nu_\infty = (14.0 \pm 0.5)$ GHz, $E_a/\kappa = 270 \pm 4$ K and $T_{VF} = 330$ K] shows the qualitative difference between both crystals: $\nu(T)$ follows the Arrhenius law in SBN-81 but a proper Vogel-Fulcher law in SBN-61. Dielectric measurements in SBN-75 31 found that the main relaxation responsible for the phase transition at $T_m \sim 310$ K has also an Arrhenius behaviour with shorter relaxation time $\tau = 5.4 \times 10^{-16}$ s ($\nu = 1.9 \times 10^{15}$ Hz), but there is another relaxation at lower temperatures with $\tau = 1.8 \times 10^{-11}$ s ($\nu = 5.6 \times 10^{10}$ Hz). This is in agreement with the central peak and the relaxation time found in Brillouin scattering32, as well as with the oscillation of tiny domain walls (of size smaller than 1 nm) found in the quasielastic neutron scattering of SBN-7024, and also with the excitation $\nu_{DW}$ we found near 10 GHz.

The temperature dependence of the permittivity of the SBN-81 crystal along the polar axis is displayed in Fig. 8a for various frequencies from 100 Hz to GHz, together with the dielectric contributions of the main excitations from the fit with eq. (2). The maximum of permittivity related to $T_m$ shifts from THz to kHz, exhibiting the relaxor character. THz data show a quite marked temperature dependence, but the contribution of the overall phonons is small ($\Delta\varepsilon_{ph} \sim 40$) and relatively temperature independent. The behaviour of the dielectric strength of $\nu_{01}$ demonstrates that $\nu_{01}$ is responsible for the dielectric anomaly at $T_m$ in the GHz–MHz range; however, its dielectric contribution is not enough to explain the permittivity at lower frequencies, where the relaxation $\nu_{02}$ emerges.
The α parameters of the three Cole-Cole relaxations are presented in Fig. 8b. This parameter estimates the deviation of a relaxation from the Debye spectrum and the width of the relaxation frequency distribution. The relaxation ν01 shows an important broadening on cooling from high temperatures and also a step in α near Tm, meaning that its character is affected by the phase transition, although its frequency does not show a remarkable anomaly. Below Tm, the excitation να1 is needed to fit the microwaves data, but, as we do not have precise results in the GHz range, its α parameter (αD) is an estimation. Nevertheless, the value αD ~ 0.3–0.4, which slightly increases towards low temperatures, corresponds well to the attribution of the νD excitation to domain walls oscillations. The α parameter of ν02 is relatively small at high temperatures (αν02 ~ 0.1), showing near Debye character, but it increases on cooling.

These results indicate the presence of several polarization mechanisms contributing to the dielectric response of SBN-81 at high frequencies. Their interplay is responsible for the complicated relaxor-ferroelectric behaviour. Each mechanism should be associated with some microscopic object in the lattice crystal. As known from literature and experiments8,21,22,24,25, several phenomena coexist in SBN: hopping of disordered cations, incommensurations in the positions of oxygen atoms, polar fluctuations, and PNDs. All of them contribute to the permittivity in different length scales and, consequently, in different frequency and temperature ranges. In comparison with SBN-61, SBN-81 has a variety of fast polarization mechanisms with different correlation lengths below the THz central mode, not only a strong relaxation with a huge slowing down15.

The low-frequency processes below 1 kHz are out of our analysis: dielectric measurements near 100 Hz already showed that the experimental data are dependent on the thermal prehistory of the crystal, the geometric shape of the sample, the measuring field, heating-cooling rates, etc.; therefore, we focused on higher frequency excitations to be able to compare these data with data taken on SBN-61. Our experimental results are complementary and in agreement with the dielectric experiments done at lower frequencies, where slow polarization mechanisms were discussed. Our relaxation ν02 is the relaxation f1 found in ref.8 just seen at higher temperatures and frequencies and the excitation νD confirms the existence of the actual ferroelectric phase transition at ~330 K. Below ν02 (or f1) there is a slower relaxation νm in the sub-Hertz range, which dominates in permittivity near the phase transition, and it was interpreted as the sideways breathing of the PNDs. These two relaxations (f1 and νm) display a dynamic scaling of the 3D Random-Field Ising model33 and evidence the occurrence of a phase transition from the relaxor state to the ferroelectric one with actual ferroelectric domains.

The evolution of the fast polarization dynamics in SBN-81 in the sub-phonon range can be summarized like this: as in other relaxors, a strong excitation in the THz range is assumed above the Burns temperature Tm. On cooling it splits into three components: an anharmonic vibration in the THz range (the central mode νD, related to anharmonic hopping of cations31), a strong GHz relaxation (ν01) associated to polar fluctuations or flipping of the forming PNDs3 and a lower-frequency relaxation ν02 related to breathing of PNDs8. On cooling, these relaxations become stronger, showing maxima of their contribution to the dielectric permittivity at different temperatures. The anharmonic hopping connected to νD softens to ~400 K and then hardens, but the relaxations ν01 and ν02 slow down further, showing the maximum of the dielectric contribution gradually at ~350 K and 330 K, respectively. Below 330 K, in the ferroelectric phase, νD appears in the GHz range and all the contributions eventually weaken. The coexistence of different polarization mechanisms gives rise to the overall dielectric dispersion, which is different from the huge and single mechanism present in SBN-6115. The onset of the phase transition occurs near 400 K and then, on cooling, several mechanisms with different correlation lengths contribute to the development of ferroelectricity. The crystal reaches finally the ferroelectric state below 330 K.

Conclusions

The high-frequency dielectric response of SBN-81 single crystal along the polar axis has been studied from 10¹ to 10¹⁵ Hz by several experimental techniques in a broad-band approach. The phase transition was revealed by far infrared spectroscopy near 330 K. No classical soft phonon mode was found, although new phonons appear below Tm, which favours the order-disorder mechanism of the phase transition. Notwithstanding, an anharmonic excitation in the THz range, a central mode, softens towards 400 K, following a linear law. Due to the presence of anharmonic disorder in the crystal, it was assigned to hopping of Sr/Ba cations inside the channels of the NbO₆ octahedra network, as in SBN-61.

Several relaxations, corresponding to fast polarization mechanisms below the phonon frequencies, play an essential role in the dielectric response of the crystal. The main contribution to the permittivity comes from a strong relaxation (ν01) present in the GHz range at high temperatures which slows down on cooling following the Arrhenius law. A second relaxation with lower frequency (ν02) slows down, as well, contributing to the permittivity mainly near Tm. Both these relaxations can be assigned to polar fluctuations, probably flipping (ν01) and breathing (ν02) of polar nanodomains. The excitation that appears below Tm around 10 GHz (νD) is associated to the development of ferroelectric microdomains. Altogether, the four mechanisms explain, above the kHz range, the ferroelectric transition in SBN-81 as well as its relaxor character, which differs from the behaviour displayed by SBN-61 and lead-based relaxors.

Methods

The SBN-81 single crystal was grown by the Czochralski method, as in ref.8. The real composition of our crystal was determined using Inductively Coupled Plasma-Optical Emission Spectroscopy and corresponds to Sr₀.₈₁Ba₀.₁₇Nb₂O₅.₉₈. The possible deviation from the nominal stoichiometry is within the experimental uncertainty and is not expected to have any significant influence on our high-frequency dielectric results or on the polarization mechanism discussed in the paper. A big bulk crystal was cut in several samples with different geometries and these were studied by means of Fourier-transform IR reflectivity, time-domain THz transmission spectroscopy (TDTTS) from 0.1 to 2.5 THz, high-frequency coaxial line technique (1 MHz–1.8 GHz), and low-frequency dielectric measurements (100 Hz–1 MHz).
IR reflectivity measurements were performed on a plate (4.5 × 5 × 0.8 mm³, with the polar axis oriented along one of the edges of the plate) using a Fourier spectrometer Bruker IFS 113v equipped with two room temperature DTGS pyroelectric detectors as well as He-cooled (1.5 K) Si bolometer. The light from a Hg lamp was polarized by a metal-mesh polarizer deposited on a thin polyethylene foil. For low temperature measurements (down to 12 K) a continuous-flow Oxford Optistat CF cryostat was used and the sample was mounted in a He gas bath, and for high temperatures (300–600 K) we used a custom-made oven. Room and high temperature spectra were measured in the range 30–1800 cm⁻¹ (1–60 THz) with a resolution of 2 cm⁻¹. Due to the presence of polarizers and optical windows the accessible frequency range at low temperatures was 30–620 cm⁻¹.

TDTTS measurements were carried out on a thin polished plane-parallel sample (4.5 × 5 × 0.05 mm³) with orientation (100) in the temperature range 10–800 K, using a polarized electromagnetic field to measure the δ[ε] spectra. A custom-made time-domain THz transmission spectrometer was used to obtain the complex dielectric response from 3 to 50 cm⁻¹ with a resolution of 0.5 cm⁻¹. An Optistat CF cryostat with Mylar windows was used for measurements down to 10 K. An adapted commercial high-temperature cell Specac P/N 5850 without windows was used to heat the sample up to 800 K.

Dielectric measurements in the high-frequency range were taken on two cylindrical samples cut from the bulk (of heights and diameters ℎ₁ = 4.0 mm, ℑ = 0.95 mm and ℎ₂ = 7.0 mm, ℑ = 0.8 mm) with the polar axis along the cylinder main axis. A computer controlled high-frequency dielectric spectrometer equipped with HP 4291B impedance analyser, a Novocontrol BDS 2100 coaxial sample cell and a Sigma System M18 chamber (temperature range 100–570 K) were used. Au electrodes were sputtered on the bases of the cylinders and the impedance of the samples was recorded on cooling at a temperature rate of 1 K/min. Results from experiments on both samples were merged.

Low-frequency dielectric measurements were performed with a Hewlett-Packard 4192A impedance analyser. Gold electrodes were added onto the faces of a plate (4.5 × 5 × 0.8 mm³) with orientation (001) and also to the ends of one of the cylindrical samples (ℎ₁ = 4.0 mm, ℑ₁ = 0.95 mm). The samples were heated and cooled in the temperature range of 80–600 K and 80–450 K, respectively, at a temperature rate of 2 K/min under a measuring field of 5 V/cm.

**Data availability statement.** All data generated or analysed during this study are included in this published article.

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**Author Contributions**

E.B. conceived the study, analysed the overall data, performed the fits and wrote the main text of the manuscript. C.K. performed the TDTTS experiment and evaluated the THz data, M.K. and V.B. performed the coaxial technique experiments and microwaves data analysis, M.S. and P.B. realized the low-frequency dielectric experiments and analysed the dielectric data, J.H. offered useful comments on results and revised the manuscript and J.D. synthesized the sample and improved the manuscript.

**Additional Information**

**Competing Interests:** The authors declare that they have no competing interests.

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