MoSe$_2$/WS$_2$ heterojunction photodiode integrated with a silicon nitride waveguide for near infrared light detection with high responsivity

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Abstract

We demonstrate experimentally the realization and the characterization of a chip-scale integrated photodetector for the near-infrared spectral regime based on the integration of a MoSe$_2$/WS$_2$ heterojunction on top of a silicon nitride waveguide. This configuration achieves high responsivity of $\sim$ 1 A W$^{-1}$ at the wavelength of 780 nm (indicating an internal gain mechanism) while suppressing the dark current to the level of $\sim$ 50 pA, much lower as compared to a reference sample of just MoSe$_2$ without WS$_2$. We have measured the power spectral density of the dark current to be as low as $\sim$ 1 x $10^{-12}$ A Hz$^{-0.5}$, from which we extract the noise equivalent power (NEP) to be $\sim$ 1 x $10^{-12}$ W Hz$^{-0.5}$. To demonstrate the usefulness of the device, we use it for the characterization of the transfer function of a microring resonator that is integrated on the same chip as the photodetector. The ability to integrate local photodetectors on a chip and to operate such devices with high performance at the near-infrared regime is expected to play a critical role in future integrated devices in the field of optical communications, quantum photonics, biochemical sensing, and more.

Introduction

Integrated photonics is a fast-developing topic of research, with many promising future applications in communications$^{1-3}$, metrology$^{4-6}$, and quantum computing$^{7-9}$. The grand vision is to integrate all the relevant devices and ultimately construct self-contained systems with their own light sources and photodetectors on a chip. Establishing such self-contained systems would solve the otherwise very laborious challenge of aligning and bonding macroscopic photonic fibers to nanophotonic chips$^{10,11}$ and make such systems easy to implement in any setup.

Integrating light sources or photodetectors on a chip is a grand challenge mostly due to material mismatch. Typically, the material needed to be used for light emission or absorption is very different from the waveguide material which is specifically chosen to be as transparent as possible for the wavelength of choice in order to reduce propagation loss. The integration of different materials is typically implemented by methods such as epitaxial growth$^{12,13}$ or direct wafer bonding of e.g. III–V semiconductors directly onto the chip$^{14-21}$. The use of each of these approaches involves significant challenges, such as atomic lattice mismatch between the different materials resulting in strain and dislocations, or the high temperatures needed for the processes, which in general, are prohibitive for other fabrication steps$^{22}$. While photodetectors can also be fabricated by evaporating absorbing materials, such as metals$^{23,24}$ and amorphous silicon$^{25}$, or by using defects states in the waveguide material$^{26}$, such devices are generally limited in their performance.

A promising remedy for this grand challenge is the use of 2D materials. These materials are easier to integrate with different target substrates, as their polymer-based
dry/wet transfer processes do not rely on any chemical reactions or the formation of atomic bonds. On top of this, their flatness and their strong inter-atomic bonds make them extremely flexible and durable, allowing them to conform to a large variety of substrate geometries. As a result, it is no surprise that in recent years there has been a multitude of research results on integrating 2D materials with different photonic systems\(^{27–31}\).

While bulk semiconductors are a fundamental building block of integrated photonics, 2D semiconductors can have a similar role as they can absorb and emit photons and their optical properties can be tuned by electrostatic charge\(^ {32,33}\), light\(^ {34}\), physical stress/tension\(^ {18}\), and engineered defect states\(^ {35}\). Transition metal dichalcogenides (TMDCs) are a family of 2D semiconductors where a unit cell of the structure consists of one transition metal atom, like molybdenum or tungsten, and two chalcogenide atoms, such as selenium or sulfur. The TMDCs become direct bandgap semiconductors when thinned down to a single monolayer, otherwise, they are indirect bandgap semiconductors. For this reason, they have been used to fabricate a variety of active photonic devices such as modulators\(^ {36}\), detectors\(^ {37,38}\), and especially light sources\(^ {39}\).

Up to now, the bulk of these works have been focused on integrated silicon photonics, operating at the telecom wavelengths\(^ {27–29,31,37,40}\), and was mostly motivated by the needs of telecommunication. However, there is a growing interest in the visible and near-infrared wavelength regime which can be supported by silicon-nitride-based integrated photonics for diverse applications in metrology, quantum computing, and biology to name a few\(^ {41}\). Examples of this can be seen in the burgeoning field of nanoscale atom–light interactions. Here, hot vapors of atomic gasses, like rubidium or cesium are interfaced with integrated photonic components, such as atomic cladded waveguides (ACWG)\(^ {42–46}\), resonators\(^ {47,48}\), and surface plasmons\(^ {49–51}\), to create novel chip-scale atomic devices. Likewise, integrating biological samples with photonic circuits can enable new and interesting applications\(^ {52,53}\). Detection of visible and near-infrared light is also promising for integrated quantum photonic devices. Color centers in silicon carbide\(^ {54}\), diamond\(^ {55}\), and hexagonal boron nitride\(^ {35}\) emit visible and near IR photons, and there is a large effort to integrate these potential qubits with nanoscale devices on a chip. Moreover, most of these applications in the visible and near IR regime require on-chip detectors with very low noise-equivalent-power (NEP), as optical saturation effects in the atoms, molecules, and qubits limit the signal powers to just a few pico- or nanowatts. A recent demonstration of TMDCs for photodetection in the visible was reported\(^ {56}\). Yet, this reported device is a photodetector (no rectification) and shows very high NEP. Clearly, a chip scale integrated rectifying photodetector (e.g., photodiode) with superior performance in the visible and near-infrared is needed.

In league with this growing effort, we hereby report an on-chip photodetector in the near-infrared band based on a van der Waals heterostructure consisting of two TMDCs, MoSe\(_2\) and WS\(_2\) integrated with silicon nitride nanowaveguides. As the MoSe\(_2\) and WS\(_2\) flakes used are respectively p- and n-type semiconductors, our photodetector is essentially a heterostructure pn-junction diode. This results in a device with excellent performance metrics, namely low dark current (~50 pA) and high peak responsivity of nearly 1 A W\(^ {–1}\). Such high-performance metrics are needed if integrated photodetectors are to be fused with, e.g. atomic vapors, where the expected signal intensities are very low due to the fast onset of optical saturation of the atomic vapors with increased probe light intensity\(^ {44}\). Our device has a time response of about 20 MHz, likely limited by the specific device geometry studied here, as the carrier dynamics predict a much faster response\(^ {37}\). Finally, to demonstrate the usefulness of our photodetector, we use it to characterize the spectral transfer function of a microring resonator that is fabricated on the same chip and coupled to the same nanowaveguide. Such integration between photodetectors and photonic elements is critical for the full realization of on-chip-integrated optical devices and systems.

**Results and discussion**

A schematic of the device concept is shown in Fig. 1a. The heterojunction photodetector consists of a bulk flake of molybdenum diselenide (MoSe\(_2\)) which is straddled over a silicon nitride waveguide implemented on the silicon chip. The MoSe\(_2\) is brought into contact with a gold electrode (drain) on one side of the waveguide, and on the opposite side, it is in contact with a single layer (SL) flake of tungsten disulfide (WS\(_2\)), which is itself contacting a second gold electrode (source).

A microscope image of the actual device is shown in Fig. 1b. The transferred single layer of WS\(_2\) is in contact with one of the gold pads, whereas the MoSe\(_2\) bulk flake of thickness of ~20 nm, is straddled on top of the second gold pad and the waveguide core region, covering ~13 μm length of the silicon nitride waveguide (SI Fig. S1). The ML-MoSe\(_2\) extends to the SL-WS\(_2\), without touching the source electrode. This way, the absorption of light propagating in the nanowaveguide is facilitated only by the bulk MoSe\(_2\). 780 nm wavelength light is launched into the waveguide by using butt coupling via a lensed fiber. The light is then absorbed in the MoSe\(_2\) multi-layers as the evanescent “tail” of the waveguide field mode spatially overlaps with the MoSe\(_2\).

We simulated the fundamental mode of our waveguide structure (0.7 μm wide by 0.25 μm-thick silicon nitride core on top of silicon oxide), with (photodetector waveguide...
mode) and without (unperturbed waveguide mode) the top layer of MoSe₂, as shown in Fig. 1c. In the unperturbed waveguide, the electromagnetic field intensity resides mostly in the center of the waveguide core (Fig. 1c top), while in the photodetector waveguide, the maximal electromagnetic field intensity is found to be at the interface between the Si₃N₄ and the MoSe₂ layers, (Fig. 1c bottom). This ensures high absorption in the MoSe₂. In terms of coupling between the unperturbed and perturbed modes, we find a coupling of 76%, meaning that 24% of the light in the unperturbed waveguide is either reflected or scattered out when interacting with the MoSe₂ layer, while the remaining 76% is rapidly absorbed by the MoSe₂ layer. From the simulation, we have extracted the propagation loss within the photodetector waveguide to be ~10⁷ dB cm⁻¹. This means that nearly 100% optical absorption (~20 dB) can be achieved with a device as short as ~2 μm.

Both TMDC layers were transferred as the last step of the device fabrication process. The flakes were transferred directly on top of the gold contacts, in order to avoid contamination and doping of the 2D materials during the lithography step used for generating the contact pads. The TMDC heterojunction is made using a standard dry viscoelastic polymer transfer method, see the section “Materials and methods” below for more details on device fabrication in general. A sketch of the device cross-section structure can be seen in Fig. 1d, where we also have defined the electrode configuration and the bias $V_{ds}$.

The MoSe₂ flake is p-doped while the WS₂ is n-doped. From observing the band diagram in Fig. 1e we can see that the heterojunction is a PN type II (staggered gap) junction. As the two conduction bands are more or less aligned, electrons see a small energy gap and can easily move between the flakes. On the other hand, the valence bands are quite largely misaligned, effectively blocking the hole transfer from the WS₂ to the MoSe₂ side of the junction. This is indeed what should make the junction behave as a current rectifier, i.e. a diode. For this reason,
we can operate the device in its reverse bias to minimize the measured dark current while still maximizing the collected photocurrent. The band structure of the heterostructure was measured by absorption spectroscopy and by Kelvin probe force microscopy (KPFM) and is shown in detail (see SI Fig. S7).

To validate the above-mentioned expectations, we have measured the IV characteristics of our device using a source measure unit (SMU), at dark and under illumination in the waveguide (see Fig. 2). For full details of the electrical and optical characterization, see the “Materials and methods” section. Indeed, we observe a clear region of forward (high dark current) and reverse (low dark current) bias, with a dark current as low as ~50 pA. This is in comparison to a reference sample, consisting of MoSe2 without WS2, where no rectification is observed, and the dark current is significantly higher ~0.5 nA (see SI Fig. S4). Additionally, we see a large photocurrent that is generated when 780 nm light is introduced into the waveguide. Interestingly, we see that the rectification slowly breaks down for increasing optical powers (marked by red-colored curves in Fig. 2a). The forward and reverse currents are nearly identical for 5 μW optical power, and for powers larger than this we even see that the forward and reverse bias actually flips. This effect is likely due to a kind of photogating effect altering the band alignment between the two materials in the junction.

Next, we plot the resulting photocurrent (total current minus the dark current) versus the optical power in the waveguide, for three representative reverse bias levels (see Fig. 2c). We note that the behavior is sub-linear, indicating optical saturation effects in our device. If, however, we restrict ourselves to the lowest powers used, we see a nearly linear regime of optical response with a responsivity of roughly 1 A W−1 for a reverse bias of 2 V (see Fig. 2c). Such high responsivity is a clear indication of a gain mechanism in our device, as an ideal photodiode would have $R = \frac{A}{\eta}$ with absorption, $A$, and quantum efficiency, $\eta$, both equal 1.

Figure 2d shows the device’s responsivity as a function of incident power. We see clearly that the responsivity decreases for larger optical powers, again indicating that optical saturation effects are non-negligible for our device for even nominal optical powers.

The heterojunction band structure diagram is shown in Fig. 2e, at forward and reverse bias. The implementation of the photodetector with an asymmetrical contact was made possible through the introduction of a WS2 monolayer between one gold contact and the bulk MoSe2, where light absorption occurs. As explained earlier, with such a device configuration, there is a large band offset in

Fig. 2 Characterization of the MoSe2-WS2 photodiode. a IV-characteristics of the diode for several optical powers in the waveguide. b Photocurrent versus incident power on the diode through the waveguide. c Photocurrent for the low power (linear optical response) region. The slope shows near 1 A W−1 responsivity for a reverse bias of 2 V. d Responsivity as a result of incident power, showing that the device saturates for even nominal powers. e Band diagram of the MoSe2-WS2 heterojunction, showing charge transfer and the hole barrier at forward and reverse bias. The error bars for (b) and (c) are multiplied by a factor of 3.
the valence band constituting a barrier for the hole carriers. This reduces the dark current without affecting the collection of photo-generated carriers. The effective energy barrier height for the dark current was extracted from a series of current vs. temperature measurements and was found to be 0.525 eV (see SI Fig. S3). This value corresponds to the energy barrier for electrons traversing from one side to the other of the device, and thus corresponds to several effects, including for example, non-Ohmic contacts, tunneling barriers, the band misalignment of the heterostructure, etc. As this barrier energy is smaller than the bandgap energy of the absorbing MoSe$_2$ layer, it thus does not limit the working range of wavelengths of our device. While the region of interest (the main area where absorption takes place) of the photodetector is the MoSe$_2$ multi-layer, the asymmetrical contact makes our device an excellent current rectifier (SI Fig. S4). We have measured several devices and the results were consistent in all of them. Data for additional devices is given in the SI.

Next, we characterize the time/frequency response and noise characteristics of the device. Figure 3a shows the setup configuration for the time/frequency response measurements. In essence, the light coupled into the nanowaveguide via the lensed fiber is first amplitude/intensity modulated by a Mach–Zehnder Modulator (MZM). By altering the modulation frequency, we can observe the device’s frequency response. For more details, see the “Materials and methods” section below. Figure 3b shows the device’s frequency response. The modulation frequency of the incident light is gradually increased until we reach the point where we exceed the response time of the photodetector. From this measurement, we found the cutoff frequency (−3 dB point) to be ~20 MHz. The solid line shown in Fig. 3b is fit to a simple RC low-pass filter model. Likewise, when modulating the device’s input light, we can measure the rise and fall times (see Fig. 3c). The rise time (defined by the time it takes the signal amplitude to increase from 10% to 90% of the maximum value), was found to be ~28 ns. Likewise, the fall time (defined by the time it takes the signal amplitude to decrease from 90% to 10% of the maximum value) was found to be ~20 ns.

In order to analyze the noise characteristics of the device, we have measured the power spectral density (PSD) of the dark current. The device’s dark current is
monitored for 8 min, and the corresponding signal is Fourier transformed to get its frequency components (see the “Materials and methods” section for full explanation). The results can be seen in Fig. 3d. From the PSD we can find the noise-equivalent-power (NEP) which is given by the PSD at 1 Hz divided by the device’s responsivity, which is roughly ~1 A/W for low powers. We thus obtain a NEP of ~1 × 10^{-12} W/Hz^{0.5}. This NEP value is comparable to many of the commercial silicon-based photodetectors and is significantly better than previously published devices.

As our photodetector is optimal for low power levels in the near-infrared, showing good NEP with decent frequency response, and given its ease of integration on the chip, it is ideal for on-chip low-power applications such as monitoring optical components and operating in a low-power regime where saturation prohibits the use of high powers (e.g. in the case of atomic cladded waveguides). As such, we now demonstrate the usefulness of our on-chip device for the characterization of a microring resonator (MRR) that is integrated on the same chip as the photodetector. Light of 780 nm wavelength was launched into the silicon nitride waveguide and coupled into the microring resonator. After interacting with the MRR, light is coupled back to the waveguide and is absorbed by the MoSe2/WS2 photodetector. The absorbed light converts to a photocurrent such that the MRR is electrically characterized by the integrated photodetector. Figure 4a shows a schematic of how the microring resonator is coupled to the same waveguide as the photodetector device. Figure 4b shows the detected photocurrent when the wavelength is scanned over a range of a few nanometers. Multiple resonances with a notch-like characteristic can easily be observed. Figure 4c is a zoom-in on a single resonance.

From the measurements, we can extract the free spectral range (FSR) of the 110 μm diameter microring resonator as 13.5 nm, which is in agreement with the theory. We can also determine the full-width-half-maximum (FWHM) for a given resonance as to be 26 pm with a corresponding Q-factor of ~30,000.

Conclusions
To conclude, we have demonstrated a chip scale photodetector with low dark current and high responsivity operating in the near-infrared spectrum that is realized by integrating a heterostructure of the 2D material TMDCs MoSe2 and WS2 with a silicon nitride waveguide. The propagation loss of the silicon nitride waveguide before transferring the 2D materials is about 3 dB cm^{-1} (see SI Fig. S8). The MoSe2 consists of several layers, which ensures very high absorption of the incident radiation propagating along the silicon nitride waveguide. By adding a monolayer of WS2 in between the MoSe2 and the second gold electrode of the device, the dark current of the device is greatly suppressed when operating in the reverse bias, resulting in a heterostructure diode. We have measured a dark current as low as ~50 pA, significantly lower as compared to a pure MoSe2 device without WS2. When operating in the linear device response regime (i.e., at low intensities below optical saturation), we were able to achieve a responsivity as high as ~1 A W^{-1} for 780 nm light, indicating a slight gain mechanism in our device. In terms of the nature of the observed gain in the device, we
can speculate that there may be a photoconductive effect related to defect states in the MoSe$_2$ working as charge traps, such that when the device is illuminated these trapped charges are released, resulting in a carrier multiplication effect. Another explanation could be related to the observed breakdown of the diode’s rectifying behavior with light in Fig. 2a. As the device is illuminated with increasing optical powers, there is some kind of ‘photogating’ effect that is breaking down the current rectification of the device, thus the current in the device is increased overall more than what is just the result of the added photocurrent. Such photogating has been observed before$^{56,60}$, and is the photonic equivalent of electrical gated rectification$^{61}$.

From frequency response measurements, we have found the 3 dB cutoff frequency to be $\sim 20$ MHz. The frequency response is most likely limited by the relatively large device geometry increasing its RC time constant. We believe that a device with a frequency response in the GHz regime may be feasible using a modified geometry. This line of thinking is based on the fact that the carrier dynamics of the materials predict much faster speeds$^{47}$. Frequency response of 20 MHz is sufficient for many switching, tuning, and monitoring applications. As for operating with Rb atoms integrated into waveguides, it was reported that the relevant bandwidth is $\sim 1$ GHz$^{45,49,62}$. As for device stability, we did not notice any major changes in the photocurrent over time (see SI Fig. S6).

By measuring the PSD of our device’s dark current we find the NEP to be as low as $\sim 1 \times 10^{-12}$ WH z$^{-0.5}$. Taking advantage of this merit, we have used our photodetector to characterize the transfer function of an optical microring resonator that is integrated on the same chip as the photodetector. Being able to provide electrical measurements on a chip, we believe that our device demonstration is another step towards realizing fully on-chip integrated photonic devices. Additionally, the general performance metrics of the detector demonstrated here should also make it fully compatible with many of the growing topics in visible and near-infrared integrated photonics, such as waveguides integrated with atomic vapors, nano-diamond color center qubits, and biological lab-on-a-chip systems.

**Materials and methods**

**Waveguide fabrication**

A 250 nm-thick layer of silicon nitride is deposited by low-pressure chemical vapor deposition (LPCVD) on 2 µm-thick thermal oxides. The waveguide cross-section is 0.7 µm width and 0.25 µm height. The waveguide geometry is defined by e-beam lithography and the structure is transferred from the resist to the silicon nitride by reactive ion etching$^{42}$. Next, the wafers are diced such that it is possible to butt couple light into the waveguide’s end facets. After this, another step of e-beam lithography is done to define the source and drain metal electrodes. The electrodes are made by evaporating 8 nm of Ti followed by 80 nm of Au using an evaporator and lift-off process.

**TMDC heterostructure fabrication**

Bulk flakes of MoSe$_2$ and WS$_2$ (HQ Graphene) are exfoliated via the Scotch tape method onto polydimethylsiloxane (PDMS) cut out from a Gel-Pak. Before exfoliation, the PDMS has been treated in an ozonator for 30 min to ensure minimal contamination from uncross-linked molecules from the PDMS. Next, the target flakes are found using an optical microscope and transferred to the intended waveguide by dry transfer technique with the help of a home-built 2D transfer setup$^{63,64}$.

**Optical characterization**

To characterize the samples’ electro-optic response, a 780 nm NewFocus TLB-67000 tunable laser source is butt coupled into the waveguide using a PM lensed fiber. Then, a Keysight B2901A SMU is connected to the gold electrodes using tungsten probes and 3D stages. Light at different power levels is injected into the waveguide, and the SMU performs a voltage sweep, $V_{ds}$ while measuring the current. To determine the laser power in the waveguide, the transmission through an identical waveguide without the 2D materials transferred on top is measured, collecting the transmitted light is with another lensed fiber, and measuring the optical power by a Newport power meter.

**Time response measurements**

The time response measurements follow the same configuration as the other optical measurements above, with the exception that the input 780 nm laser is modulated by an iXblue Mach–Zehnder modulator (MZM). A sine RF signal is supplied to the modulator by a Keysight DSOX1004A digital storage oscilloscope, and the modulated signal is again butt coupled into the target waveguide. The modulated photoresponse from the heterostructure detector is then monitored on a Zurich Instruments digital lock-in amplifier oscilloscope.

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**Author contributions**

The project was conceived by all authors. Waveguide fabrication was done by R.Z. TMDC Heterostructures were fabricated by S.R.K.C.I. Optical and time response measurements were done by R.G., S.R.K.C.I, and R.Z. Data analysis was done by R.G. and C.F. U.L supervised the project. All authors contributed to the writing of the manuscript and the interpretation of the results. R.G., S.R.K.C.I contributed equally to the manuscript.

**Conflict of interest**

The authors declare no competing interests.
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