Aromatic Polyesters Containing Ether and a Kinked Aromatic Amide Structure in the Main Chain: Synthesis and Characterisation

Satish S. Deokar 1,* , Makarand D. Joshi 2 , Asiya M. Tamboli 3,* , Sultan Alshehri 4 ,* , Mohammed M. Ghoneim 5 ,* , Nguyen Tam Nguyen Truong 3,* , Mohaseen S. Tamboli 6 ,* and Noormahamad N. Maldar 7

Abstract: A novel bisphenol containing preformed multiple ether and amide linkages, N, N′-Bis (4-hydroxyphenoxyphenylene) isophthalamide (IPCD), was prepared and analysed by spectroscopic methods. New aromatic polyesters were prepared by polycondensation of IPCD with 1, 3-benzene dicarboxylic acid chloride and/or 1, 4-benzene dicarboxylic acid chloride. These obtained polyesters were structurally analysed by infra-red spectroscopy, measurements of inherent viscosity, wide-angle X-ray diffraction patterns, and thermal techniques such as differential scanning calorimetry (DSC), thermogravimetric analysis (TGA), and solubility tests in organic solvents. The synthesised polyesters had high molecular weights, as indicated by their inherent viscosities from 0.67 to 0.78 dL/g in N-methyl-2-pyrrolidone. The incorporation of ether and kinked aromatic amide moieties in the main polyester chains greatly affected the properties of these aromatic polyesters. The prepared polyesters readily dissolved in amide-type polar aprotic solvents and pyridine, indicating their solution processability. The DSC curves above the polyesters showed glass transition temperatures of 194 to 269 °C. TGA indicated that these newly obtained polyesters were stable up to 301 °C and retained a 39 to 48% weight at 900 °C. W-XRD analyses showed that the newly synthesised polyesters were amorphous, which is reflected in their solubility behaviour.

Keywords: N, N′-Bis (4-hydroxyphenoxyphenylene) isophthalamide; soluble aromatic polyesters; glass transition temperatures; high-temperature stability; XRD

1. Introduction

Aromatic polyesters have good thermal stability and other physico-mechanical properties, leading to their wide applications in various important state-of-the-art industries [1–3]. However, wholly polyarylates are insoluble and have high glass transition or melting temperatures and are hence difficult to process. The ordering and structural variations of polymer backbones decide the final properties of the polymer [4,5]. Earlier investigations showed that modified cardo groups [6,7] and pendant moieties [8–12] produced amorphous polyesters that dissolved in organic solvents. Polyesters derived from CNSL-based monomers showed relatively lower glass transition temperatures in addition to the enhanced solubility of these polyesters in some organic solvents, probably due to the plasticising effect of the pendant long pentadecyl alkyl groups [5,9]. Good melt-processability
of the polymers required lower glass transition temperatures so that processing could be performed at relatively lower temperatures. However, the same pentadecyl alkyl groups decreased the thermal stability of the resulting polyesters. [11] Pendant biphenyl [3], naphthyl [10], or phenothiazine moiety [12] in aromatic polyesters improved the solubility of resulting polyesters, and there was no significant loss in their thermal stability. Linkages such as ether and sulphone in polymers increased the solubility and solution processability without any comprise in the polymer thermal stability [13–16].

Aromatic polymers containing the ether and amide side groups showed improved solubility and thermal stability [17]. To achieve these properties in high-performance polymers, various flexible ether linkages were incorporated to yield poly(ether ester)s and poly(ether amide)s [18,19]. High-molecular-weight terephthalate copolyesters from flexible cycloaliphatic diols or diacids displayed a coherent structure-property relationship related to the rigid, flexible, and bulky structures [20,21]. The copolyesterification process also enhanced the solubility of resulting polyesters due to a formation of the random, irregular arrangement in polymer chains. Thus, ether moieties in aromatic polymers and random copolymerisation increased the flexibility of these polyesters. Enhanced flexibility decreased glass transition temperatures and enhanced polymer solubility. Thus it assisted in overcoming the difficulty of polymer processing [22–24].

Based on these concepts, it was thought to synthesise bisphenol-containing, preformed ether and amide groups, wherein kink or m-catenation links could be provided at few aromatic units as it would be more effective for improving the solubility of the resulting poly(ether ester)s. Thus, this report includes the synthesis and characterisation of a new, modified bisphenol, namely N,N′-[4′-(4″-hydroxyphenoxy) phenylene] isophthalamide (IPCD), with preformed flexible ether groups and kinked aromatic amide structures. IPCD was utilised to obtain novel, thermally stable, soluble and solution-processable aromatic polyesters and copolyesters. The thermal and structural characterisation of these polyesters is reported.

2. Experimental

2.1. Materials

All the chemicals were acquired from different companies and utilised exactly as they were provided. Hydroquinone (Loba make, Mumbai, India), p-Chloronitrobenzene, (Spectrochem, Bombay, India), N,N-dimethylformamide, terephthaloyl dichloride (TPC), and isophthaloyl dichloride (IPC) were purified by standard procedures [25]. Potassium carbonate (s d fine) was dried at 100 °C for 5 h and stored in a desiccator until use. Methylene dichloride and absolute ethanol were used as received.

Measurements

Synthesised new bisphenol and intermediate organic compounds and polyesters were scanned for FT-IR spectra on a Nicolet 520 FT-IR spectrometer (Thermo Fisher Scientific, Milan, Italy) using KBr pellets. 1H NMR (400 MHz) and 13C NMR (100 MHz) spectra were recorded on a Bruker Advance spectrometer (Bruker, Bremen, Germany) at 25 °C using CDCl3 and DMSO-d6 as solvents and tetramethyrsilane as a reference. Mass spectrum was recorded on the Bruker LCMS/MS (QTOF) instrument (Bruker, Bremen, Germany) using an electrospray ionisation (ESI) technique. Solubility of polyesters was determined by dissolving 10 mg polyester in 1 mL solvent at room temperature for 24 h. Inherent viscosities of aromatic polyesters were measured with a polyester concentration of 0.5 g/dL in N-methyl-2-pyrrolidone (NMP) solvent at 30 °C under nitrogen. Thermogravimetric analysis (TGA) was recorded on a Mettler Toledo STAR instrument (Mettler Toledo DSC STAR, Zurich, Switzerland) at a heating rate of 10 °C/min. under nitrogen. Wide-angle X-ray diffraction (WAXD) was recorded with a
Rigaku X-ray diffractometer with CuKα radiation (with a wavelength = 1.5418 Å) on an angular range of 20 = 5° to 60° with 1.0°/min. scan increment at room temperature using polyester powder.

2.2. Synthesis of Bisphenol: N, N’-[4’-(4”-Hydroxyphenoxy)phenylene]isophthalamide

(a) Synthesis of 4(4’-nitrophenoxyl)phenol (NPP):

In a 500 mL one-neck, round bottom flask, hydroquinone (0.164 mol, 18.04 g), p-chloronitrobenzene (0.1148 mol, 18.08 g), potassium carbonate (0.164 mol, 22.63 g), and N,N dimethylformamide (150 mL) were stirred overnight at 30 °C and then heated at 100 °C for 5 h. Afterwards, cooling contents were mixed with cold, 5% aq. hydrochloric acid to precipitate product. The product was separated by filtration, washed with water, and dried. The product was ethanol-extracted, and the extracts were concentrated. When almost all solvent was removed, the residue was dried under vacuum. Recrystallisation of the residue from aq. ethanol produced a dark yellow pure product (NPP). Yield: 13.19 g (49.7%) and M.P.167–169 °C.

(b) Synthesis of 4-(4’ aminophenoxy)phenol (APP):

In a 250 mL three-neck, round-bottom flask NPP (0.044 mol, 10.164 g), 10% Pd/C (0.4 g) and 150 mL ethanol were stirred at 80 °C. Over a period of 1 h, hydrazine hydrate (99%, 20 mL) was added drop-wise to the reaction mixture. Refluxing was continued for 4 h. The reaction mixture was filtered to remove the Pd/C. Distillation of the filtrate was performed to recover about 125 mL of ethanol. About 150 mL of water was added to the residue in the reaction flask. Precipitate was collected by filter-funnel, washed with water, and dried. Recrystallisation of product from boiling water gave pure APP. Yield: 6.3 g (71.3%) and M.P.152–153 °C.

(c) Synthesis of N, N’-[4’-(4”-hydroxyphenoxy)phenylene]isophthalamide (IPCD):

In a 100 mL three-neck, round-bottom flask, APP (0.402 g, 2 mmol) was dissolved in DMAC (4.4 mL) under nitrogen atmosphere and the solution was cooled to −15 °C. To the solution of IPC (0.203 g, 1 mmol) four lots was added. Stirring was continued for 2 h at 0 °C and then for 12 h at 30 °C. The product was precipitated by mixing water (60 mL) with the reaction solution. Precipitate was filtered, washed with methanol, and dried (Scheme 1): yield 0.806 g (94.9%); M.P. 290 °C. Mass: m/z 533 (M + 1) ion peak.

Scheme 1. Synthesis of N, N’-[4’-(4”-hydroxyphenoxy)phenylene]isophthalamide, IPCD.
2.3. Polymerisation

Scheme 2 illustrates interfacial method of polymerisation for new bisphenol IPCD with aromatic dicarboxylic acid chloride. Procedure adopted to synthesise aromatic polyester was as follows.

**Scheme 2.** Synthesis of polyesters.

Synthesis of polyester (PE-5).
A 100 mL three-neck, round-bottom flask was charged with IPCD 0.532 g (1 mmol), NaOH 0.080 g (2 mmol), benzyl triethyl ammonium chloride 0.100 g, and water 12.5 mL and stirred at 0 °C. IPC 0.203 g (1 mmol) in 8 mL DCM was introduced into the flask in one lot. Then, the reaction mixture was stirred vigorously for an hour at 30 °C. The viscous solution was poured into excess hexane to precipitate the polyester, PE-5. The polyester was filtered, washed with distilled water, and dried. Yield: 98%, inherent viscosity, 0.67 dL/g.

Other polyesters PE-1 to PE-4 were synthesised by similar procedure.

3. Result and Discussion

The aim was to prepare modified, soluble, and thermally stable aromatic polyesters and study their thermal as well as structural characterisation.
3.1. Bisphenol, N, N’-{4′-(4″-Hydroxyphenoxy)phenylene}isophthalamide Synthesis

The bisphenol IPCD was synthesised in three steps, including hydroquinone on reaction with p-chloronitrobenzene gave 4(4′-nitrophenoxy)phenol (NPP) which, on reduction, gave 4(4′-aminophenoxy)phenol (APP). This APP (2 mol) on reaction with IPC (1 mol) gave the desired bisphenol IPCD in, overall, about 34% yields for three steps.

IPCD was analysed by FT-IR, $^1$H and $^{13}$C NMR spectroscopy, and mass spectrometry. The FT-IR spectrum of IPCD showed a broad absorption band at 3284 (O–H), 1694 (C=O) stretching, 3132, and 3047 cm$^{-1}$ (–N–H). The $^1$H NMR spectrum of IPCD showed a singlet at 10.65 corresponding to the phenolic proton and an 8.77 singlet due to amide (–NH), whereas aromatic protons appeared from 8.33 to 7.15 δ (Figure 1).

Figure 1. $^1$H NMR spectrum of IPCD.

Figure 2 shows a $^{13}$C NMR spectrum of IPCD. It exhibited 13 different peaks for 13 different non-equivalent carbons. An amide carbonyl peak appeared at 166, and aromatic carbon peaks were in the range of 116 to 154 δ. A $^{13}$C DEPT NMR spectrum of IPCD confirmed the structure, wherein quaternary carbons were absent (Figure 3). Mass spectrometry of IPCD exhibited a peak at m/z = 533 units due to the (M + 1)$^+$ peak, confirming the formation of IPCD as per the assigned chemical structure.
Figure 2. $^{13}$C NMR spectrum of IPCD.

Figure 3. DEPT $^{13}$C NMR spectrum of IPCD.

3.2. Polymerisation

A series of aromatic polyesters (PE-1 to PE-5) was prepared by utilising an interfacial polymerisation method wherein bisphenol IPCD was reacted with aromatic dicarboxylic acid chloride. The polymerisation was performed at ambient temperature for an hour. The
polymerisation proceeded smoothly, giving a viscous solution. Polyester was precipitated by pouring the reaction solution into a large volume of hexane.

The inherent viscosities and the yields of these polyesters from IPCD and diacid chlorides are given in Table 1.

Table 1. Inherent viscosities and percentage yields of polyesters a (PE-1 to PE-5) from IPCD.

| Sr. No. | Polyester Code | mol%  | Yield (%) | \( \eta_{inh} \) (dL/g) |
|---------|----------------|-------|-----------|-------------------------|
| 1       | PE-1           | 100   | 00        | 96                      | 0.78                    |
| 2       | PE-2           | 75    | 25        | 97                      | 0.75                    |
| 3       | PE-3           | 50    | 50        | 97                      | 0.73                    |
| 4       | PE-4           | 25    | 75        | 96                      | 0.69                    |
| 5       | PE-5           | 00    | 100       | 98                      | 0.67                    |

a Polymerisation was performed using 1 mmol each of IPCD and acid chloride. b Measured at 0.5% polyester concentration at 30 °C.

3.3. Characterisation of Polyesters

High molecular weight polyesters were obtained as indicated by their inherent viscosities ranging from 0.67 to 0.78 dL/g in NMP Polyester PE-1 derived from TPC, which showed the highest inherent viscosity among all five polyesters studied, probably due to the higher reactivity of terephthalic acid chloride.

Analysis of polyesters by FT-IR spectroscopy confirmed the formation of the desired polyesters. The FT-IR spectrum of polyester PE-5 is given in Figure 4. The characteristic absorption bands at 1740 cm\(^{-1}\) corresponding to newly formed ester carbonyl (C=O stretching) of the polyester, and absorption bands of the preformed amide group at 3267 cm\(^{-1}\) (due to the –NH stretching) and at around 1672 cm\(^{-1}\) (due to the carbonyl stretching of amide), were seen in the FT-IR spectrum [3,8]. The absence of an absorption band at 3491 cm\(^{-1}\) indicated that all hydroxyl groups of bisphenol reacted completely to yield polyester.

Figure 4. FT-IR spectrum of PE-5.
Solubility of Polyesters:

Synthesised aromatic polyesters containing multiple ether and amide groups were tested qualitatively for their solubility in various solvents. Polyesters were considered as ‘soluble’ when tested by dissolving 10 mg of the polyester in 10 mL of the solvent to give a 1% polymer solution at 30 °C. All these aromatic polyesters were soluble in polar aprotic solvents and in pyridine.

The aromatic polyesters showed good solubility. This may be due to the incorporation of multiple ether groups. This modification decreased intermolecular interactions between polyester chains, decreased the packing efficiency, and produced the amorphous soluble polyesters. Further m-catenation in the kinked aromatic amide structure in these polyesters favoured better solubility than if it were a p-catenated aromatic amide structure.

All these polyesters (PE-1 to PE-5) were soluble in organic solvents, irrespective of the structure of the diacidchlorides, which indicated that solubility was mainly due to the multiple ether moieties and the kinked 1, 3-phenylene amide structure. These soluble polyesters may find varieties of applications as high-performance polymers in the electronics, automotive, textile, and aerospace industries as films, membranes, coatings, varnishes, adhesives, paints, spun fibres, etc.

Thermal properties of polyesters:

Polyesters were evaluated for the thermal properties by DSC (Figure 5) and TGA (Figure 6) analyses. The glass transition temperature \( T_g \) and initial decomposition temperature \( T_i \), the temperatures of the maximum rate of decomposition \( T_{\text{max}} \), and char yields at 900 °C of (PE-1 to PE-5) are presented in Table 2. Peaks corresponding to melting were not observed in the DSC curves, indicating that polyesters are amorphous. Polyesters PE-1 to PE-5 showed different \( T_g \)s from 194 °C to 269 °C depending on the structure and stiffness of the diacidchlorides, which decided the rigidity of these polyesters. It was observed that the \( T_g \) value for PE-5, which was derived from m-catenated IPC, was lower (194 °C) than the \( T_g \) values for the remaining four polyesters, PE-1 to PE-4. Aromatic polyester PE-1, which was derived from p-catenated TPC, showed the highest \( T_g \) (269 °C) due to the 1, 4 aromatic linkages. The \( T_g \) values of polyesters increased stepwise as the TPC content in the polyesters increased from PE-5 to PE-1, as rigidity increased in diacidchlorides component.

| Sr. No. | Polyester Code | \( T_i \) (°C) | \( T_{\text{max}} \) (°C) | Residual wt. at 900 °C (%) | \( T_g \) (°C) |
|---------|----------------|----------------|-----------------------------|-----------------------------|-------------|
| 1       | PE-1           | 325            | 542                         | 48                          | 269         |
| 2       | PE-2           | 320            | 530                         | 45                          | 223         |
| 3       | PE-3           | 315            | 525                         | 44                          | 208         |
| 4       | PE-4           | 309            | 521                         | 42                          | 198         |
| 5       | PE-5           | 301            | 503                         | 39                          | 194         |

Flexible ether moieties in the main chain favoured polyester chain movements, and the m-catenated aromatic amide structure in the main chain broke the linearity of the aromaticity. The combined effect of these two modified units in the polyesters resulted in the lowering of the glass transition temperatures. On the other hand, amide groups in the polyester chains may promote slightly increased \( T_g \) values. Thus, there is an overall effect of the ether as well as amide units. However, the \( T_g \) values of these aromatic polyesters were not as high as those of the wholly aromatic polyester, Ekonol, poly(p-benzoate) \( (T_g > 500 \, ^\circ \text{C}) \).
Coatings 2022, 11, x FOR PEER REVIEW 9 of 13

(194 °C) than the Tg values for the remaining four polyesters, PE-1 to PE-4. Aromatic poly-
yester PE-1, which was derived from p-catenated TPC, showed the highest Tg (269 °C) due

Flexible ether moieties in the main chain favoured polyester chain movements, and

The combined effect of these two modified units in the polyesters resulted in

Tg values of these aromatic polyesters were not as high as those of the wholly aromatic polyester, Ekonol, poly(p-benzoate) (Tg >500 °C).

Figure 5. DSC curves of PE-1 to PE-5.

Table 2. Thermal analysis of polyesters (PE-1 to PE-5).

| Sr. No. | Polyester Code | Ti (°C) | Tmax (°C) | Residual wt. at 900 °C (%) | Tg (°C) |
|---------|---------------|---------|-----------|--------------------------|---------|
| 1       | PE-1          | 325     | 542       | 48                       | 269     |
| 2       | PE-2          | 320     | 530       | 45                       | 223     |
| 3       | PE-3          | 315     | 525       | 44                       | 208     |
| 4       | PE-4          | 309     | 521       | 42                       | 198     |
| 5       | PE-5          | 301     | 503       | 39                       | 194     |

X-ray Diffraction:

XRD of these aromatic polyesters containing multiple ether and amide links was per-
formed for 2θ = 5–60° at ambient temperature. Figure 7 shows broad hallow peaks with a

The kinked 1, 3-phenylene amide structure and abundant, flexible ether link-
ages in the main polyester chain may be responsible for producing the amorphous poly-
esters. Obviously, the improved solubility of these polyesters can be assigned to the com-
pletely amorphous nature of these aromatic polyesters and copolyesters.

Figure 6. TGA curves of PE-1 to PE-5.

Figure 5. DSC curves of PE-1 to PE-5.

Figure 6. TGA curves of PE-1 to PE-5.
Aromatic polyesters PE-1 to PA-5 showed no weight loss below 301 to 325 °C, and the $T_{\text{max}}$ for these aromatic polyesters was from 503 to 542 °C. The residual weights (%) at 900 °C for these aromatic polyesters were from 39% to 48%. All these thermal parameters indicated that the polyesters were highly thermally stable. Large amounts of the aromatic structures and thermally stable amide and ether moieties in the main chain of these polyesters may be prime contributing factors towards good thermal stability.

**X-ray Diffraction:**

XRD of these aromatic polyesters containing multiple ether and amide links was performed for $2\theta = 5$–$60^\circ$ at ambient temperature. Figure 7 shows broad hallow peaks with a slight shoulder at $2\theta = 10$–$35^\circ$ and halos were centred at about 22°, with a minor shoulder holo at about 15°. Polyesters were amorphous, as the XRD diagram did not show any very sharp peaks. The kinked 1, 3-phenylene amide structure and abundant, flexible ether linkages in the main polyester chain may be responsible for producing the amorphous polyesters. Obviously, the improved solubility of these polyesters can be assigned to the completely amorphous nature of these aromatic polyesters and copolyesters.

![Figure 7. XRD curves of PE-1 to PE-5.](image)

**3.4. Conclusions**

The synthesis of new bisphenol containing two ether moieties and kinked aromatic amide, N, N’-[4’-(4’-hydroxyphenoxy)phenylene]isophthalamide (IPCD), was performed, and the bisphenol was completely analysed by spectroscopic methods. Thermally stable and soluble aromatic polyesters containing ether groups and kinked 1, 3-phenylene amide structure were designed and synthesised by an interfacial method from IPCD with two isomeric aromatic dicarboxylic acid chlorides. The molecular weights of the synthesised polyesters were high, as illustrated through the range of inherent viscosities from 0.67 to 0.78 dL/g in NMP. Incorporation of the ether groups and kinked 1, 3-phenylene amide
structure in the main polyester chains resulted in soluble polyesters. Polyesters were dissolved in organic aprotic polar solvents and pyridine. Thus, the obtained soluble aromatic polyesters had good thermal stability and successfully overcame the solution processability issues of wholly aromatic polyesters. These polyesters, when characterised for thermal properties, showed glass transition temperatures from 194 to 269 °C. There was a clear increasing trend in the values of $T_g$ as the content of TPC increased. Thermogravimetric analysis of these thermally stable aromatic polyesters showed no weight loss below 301 °C, whereas the $T_{\text{max}}$ was more than 503 °C and char yields at 900 °C were from 39% to 48%. All these prepared polyesters, when studied by WXRD, showed an amorphous nature because the introduction of multiple ethers and a kinked 1, 3-phenylene amide structure may have disrupted the polymer chain packing. These thermally stable, soluble aromatic polyesters could be solution-processable. Thus, studies showed that introducing an ether and m-catenated phenylene structure may be useful to tailor-make $T_g$ of aromatic high-performance polyesters.

**Author Contributions:** Writing—original draft, conceptualisation, methodology, and data interpretation, S.S.D.; data curation, validation, and resources, M.D.J.; writing—review and editing, A.M.T.; funding acquisition, S.A. and M.M.G.; visualisation and formal analysis, N.T.N.T.; writing—review and editing, M.S.T.; supervision and project administration, N.N.M. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research received no external funding.

**Institutional Review Board Statement:** Not applicable.

**Informed Consent Statement:** Not applicable.

**Data Availability Statement:** Not applicable.

**Conflicts of Interest:** The authors declare no conflict of interest.

**References**

1. Imai, Y.; Maldar, N.N.; Kakimoto, M. Synthesis and characterization of soluble aromatic polyazomethines from 2,5-bis(4-aminophenyl)-3,4-diphenylthiophene and aromatic dialdehydes. *J. Polym. Sci. Polym. Chem.* 1984, 22, 3771–3778. [CrossRef]

2. Joshi, M.D.; Lonikar, S.V.; Maldar, N.N.; Wadgaonkar, P.P. Synthesis and Characterization of Phenylated Poly (Phenylquinoxaline)s. *J. Macromol. Sci. Part A* 1997, 34, 1077–1083. [CrossRef]

3. Honkhambe, P.N.; Avadhani, C.V.; Wadgaonkar, P.P.; Salunkhe, M.M. Synthesis and characterization of new aromatic polyesters containing biphenyl side groups. *J. Appl. Polym. Sci.* 2007, 106, 3105–3110. [CrossRef]

4. Jadhav, A.S.; Maldar, N.N.; Shinde, B.M.; Vernekar, S.P. Synthesis and characterization of silicon containing polyamides from aromatic sulfone ether diamines and aromatic organosilicon diacid chlorides. *J. Polym. Sci. Polym. Chem.* 1991, 29, 147–153. [CrossRef]

5. Jadhav, A.S.; Vernekar, S.P.; Maldar, N.N. Synthesis and characterization of new aromatic sulfone ether polyamides containing pendant pentadecyl groups. *Polym. Int.* 1993, 32, 5–11. [CrossRef]

6. Vibhute, S.S.; Joshi, M.D.; Wadgaonkar, P.P.; Patil, A.S.; Maldar, N.N. Synthesis and characterization of new cardo polyesters. *J. Polym. Sci. Polym. Chem.* 1997, 35, 3227–3234. [CrossRef]

7. Shigte, R.D.; Chatterjee, D.; Tawade, B.V.; Shrimant, B.; Wadgaonkar, P.P. Aromatic polyesters containing cardo perhydrocycumyl cyclohexylidene groups: Synthesis, Characterization and gas permeation study. *J. Macromol. Sci. Part A* 2019, 56, 136–145. [CrossRef]

8. Bapat, S.P.; Jadhav, S.A.; Valsange, N.G.; Tawade, B.V.; Honkhambe, P.N.; Chavan, N.N.; Wadgaonkar, P.P. Aromatic polyesters containing pendent 4-(phenylsulfonyl)phenyl groups: Synthesis and characterization. *J. Polym. Res.* 2017, 24, 57. [CrossRef]

9. Tawade, B.V.; Salunke, J.K.; Sane, P.S.; Wadgaonkar, P.P. Processable aromatic polyesters based on bisphenol derived from cashew nut shell liquid: Synthesis and characterization. *J. Polym. Res.* 2014, 21, 617. [CrossRef]

10. Honkhambe, P.N.; Biyani, M.V.; Bhairamadgi, N.S.; Wadgaonkar, P.P.; Salunkhe, M.M. Synthesis and Characterization of New Aromatic Polyesters Containing Pendent Naphthyl Units. *J. Appl. Polym. Sci.* 2010, 117, 2545–2552. [CrossRef]

11. More, A.S.; Naik, P.V.; Kumbhar, K.P.; Wadgaonkar, P.P. Synthesis and characterization of polyesters based on 1,1-[bis(4-hydroxyphenyl)-4’-pentadecylphenyl]ethane. *Polym. Int.* 2010, 59, 1408–1414. [CrossRef]

12. Ma, X.; Niu, H.; Cai, W.; Xiao, T.; Wang, C.; Bai, X.; Zhang, Y.; Wang, W. New high-solubility aromatic polyesters with pendent phenothiazine: Synthesis, electrochromic and optoelectronic properties. *React. Funct. Polym.* 2016, 108, 63–70. [CrossRef]

13. Basutkar, P.H.; Joshi, M.D.; Lonikar, S.V.; Maldar, N.N.; Idage, B.B. Synthesis and characterization of phenylated aromatic poly(amide–amide)s. *J. Appl. Polym. Sci.* 1998, 68, 1523–1530. [CrossRef]
14. Ubale, V.P.; Sagar, A.D.; Maldar, N.N.; Birajdar, M.V. Synthesis and characterization of aromatic–aliphatic polyamides. *J. Appl. Polym. Sci.* 2001, 79, 566–571. [CrossRef]

15. Pal, R.R.; Patil, P.S.; Salunkhe, N.N.; Maldar, N.N.; Wadgaonkar, P.P. Synthesis and characterization of aromatic polyamides containing an s-triazine ring with thiophenoxy linkages. *Polym. Int.* 2005, 54, 569–575. [CrossRef]

16. Patil, V.B.; Medhi, M.; Bhairamadgi, N.S.; Wadgaonkar, P.P.; Maldar, N.N. Synthesis and characterization of polyesters from 2, 3-bis (4′-hydroxy phenyl) quinoxaline and 2, 3-bis (2′-hydroxynaphthalene-6-yl) quinoxaline. *Mater. Sci. Eng. B* 2010, 168, 186–192. [CrossRef]

17. Brianna, M.; Upton, B.M.; Andrea, M.; Kasko, A.M. Biomass-Derived Poly(ether-amide)s Incorporating Hydroxycinnamates. *Biomacromolecules* 2019, 20, 758–766.

18. Latha, G.; Murugavel, S.C. Synthesis and characterization of cardo poly (ether-ester)s containing 1, 1-bis (4-hydroxyphenyl) cyclohexane in the main chain. *Polym. Bull.* 2016, 73, 3237–3253. [CrossRef]

19. Shibasaki, Y.; Mori, T.; Fujimori, A.; Jikei, M.; Sawada, H.; Oishi, Y. Poly(amide–ether) Thermoplastic Elastomers Based on Monodisperse Aromatic Amide Hard Segments as Shape-Memory and Moisture-Responsive Materials. *Macromolecules* 2018, 51, 9430–9441. [CrossRef]

20. Iyer, K.A. Chain mobility, secondary relaxation, and oxygen transport in terephthalate copolyesters with rigid and flexible cyclic diols. *Polymer* 2017, 129, 117–126. [CrossRef]

21. Chi, D.; Liu, F.; Hao, C.; Chen, J.; Na, H.; Zhu, J. Identification of side chain effect as an important factor influencing the secondary relaxation of polyesters containing cyclohexylene ring. *J. Mater. Sci.* 2018, 53, 6234–6250. [CrossRef]

22. Patil, P.S.; Pal, R.R.; Salunkhe, M.M.; Maldar, N.N.; Wadgaonkar, P.P. Synthesis of aromatic poly (amide-imide) s from novel diimide-diacid (DIDA) containing sulphone and bulky pendant groups by direct polycondensation with various diamines. *Eur. Polym. J.* 2007, 43, 5047–5054. [CrossRef]

23. Pal, R.R.; Patil, P.S.; Salunkhe, M.M.; Maldar, N.N.; Wadgaonkar, P.P. Synthesis, characterization and constitutional isomerism study of new aromatic polyamides containing pendant groups based on asymmetrically substituted meta-phenylene diamines. *Eur. Polym. J.* 2009, 45, 953–959. [CrossRef]

24. Wu, Z.F.; Zhang, G.; Yan, G.M.; Lu, J.H.; Yang, J. Aromatic polyesters containing different content of thioether and methyl units: Facile synthesis & properties. *J. Polym. Res.* 2018, 25, 170–181.

25. Perrin, D.D.; Armarego WL, F.; Perrin, D.R. *Purification of Laboratory Chemicals*; Pergamon Press: Oxford, UK, 1980.