Annealing and Normalizing of AISI 1045 Steel: A Lamellae Analysis

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Abstract
Two heat treatments, annealing and normalizing were conducted to verify the alteration of the microstructure and corresponding mechanical properties of AISI 1045 steel. The effect of cooling rate in the heat treatment process on the lamellae microstructure (lamellar thickness) and hardness it was studied. Results show reduced lamellar thickness of 0.175 µm in the normalized condition (high cooling rate-fine pearlite) than in the annealed condition of 0.294 µm (low cooling rate-coarse pearlite). Using the Vickers Hardness Tester, it was determined that the hardness of AISI 1045 steel in the annealed condition and normalized condition is 56.42 HV and 202 HV, respectively.

Keywords
Hypoeutectoid steel, Phase transformations, Heat treatment, Normalizing, Annealing, Pearlite, Lamellar thickness

Introduction
Several of the various microstructures that may be produced in steel alloys depends on both the carbon content and heat treatment. Mechanical properties of metals can be altered by a number of techniques. One of the most convenient methods to introduce phase changes is heat treatment as the cooling rate has an influence in the resultant phase. Heat treatment refers to several types of heating and cycles performed on a metal to beneficially change its properties. They operate by altering the basic microstructure of the metal, which in turn determines mechanical properties [1]. Of all binary alloy systems, the one that is possibly the most important is that for iron and carbon. The development of a set of desirable mechanical characteristics in steels are accomplished by an appropriate heat treatment. Steel varies its phases depending on composition and temperature. A variety of phase transformations are important in the processing of materials, and usually they involve some alteration of the microstructure. The absolute layer thickness of the ferrite and cementite depends on the temperature (and cooling rate) at which the transformation is allowed to occur. At temperatures just below the eutectoid, relatively thick layers of both the α-ferrite and Fe₃C phases (i.e., the coarse pearlite) are produced (See Figure 1). At these temperatures, diffusion rates are relatively high, such that carbon atoms can diffuse relatively long distances, which results in the formation of thick lamellae [2].
With decreasing temperature, the carbon diffusion rate decreases, and the layers become progressively thinner. The thin-layered structure produced in the vicinity of 540 °C is termed fine pearlite.

The continuous cooling transformation (CCT) diagram shown on Figure 1, is a plot of temperature versus the logarithm of time for a steel alloy of definite composition. Used to indicate when transformations occur as the initially austenitized material is continuously cooled at a specific rate; in addition, the final microstructure and mechanical characteristics may be predicted. The CCT diagram describes more efficiently the resultant phases of normalizing and annealing. Figure 1 indicates that for a moderately rapid cooling (normalizing), the resultant phase is fine pearlite in which the alternating ferrite and cementite layers are relatively thin. Likewise, in full anneal (i.e., a slow cooling curve), coarse pearlite is formed. In coarse pearlite, the alternating ferrite and cementite layers are relatively thick. As the cooling rate decreases, the thickness and interlamellar spacing increases. Pearlite coarsening is part of the heat treatment with slow cooling that reflects higher nucleation and growth.

Heat treatments are related with cooling rates (See Figure 1 and Figure 2). Annealing is the process in which a material is heated to a critical temperature $T_c$ for an extended period of time and then

**Figure 1:** Moderately rapid and slow cooling curves superimposed on a continuous-cooling transformation diagram for eutectoid steel (after Ref. [2]).

![Figure 1: Moderately rapid and slow cooling curves superimposed on a continuous-cooling transformation diagram for eutectoid steel (after Ref. [2]).](attachment:image.png)
both temperature and time dependence of the transformation and are helpful in the determination of the resulted phase. Cooling hypoeutectoid steel first nucleates proeutectoid ferrite preferentially at austenite grain corners, followed by edges and then boundaries [3]. In this work, a discussion of the dependence of mechanical properties on lamellar thickness in the annealing and normalizing heat treatment processes is presented.

Results and Discussion

The mechanical properties are directly related to the microstructure, which in turn depends on the cooling rate experienced by the alloy. In consequence, as the cooling rate differs in the annealing and in the normalizing process, the hardness will adjust as well. A lot of work had been done to study the lamellae microstructure formed during an isothermal transformation in eutectoid steels. Contrary, a few works has been done to study the lamellae microstructure formed in continuous cooling transformation in hypoeutectoid steels. In annealing, particle growth occurs by long-range atomic diffusion [2]. Consequently, the growth rate is determined by the rate of diffusion which depends on the temperature. Moreover, the time dependence of rate, or kinetics, is important in the heat treatment since the reaction of fraction that has occurred is a function of time while temperature is maintained constant. Isothermal and continuous cooling transformation diagrams relate

Figure 2: The heat treatment process. The cooling rate for the normalizing is calculated as 4 °C/min and for the annealing as 1 °C/min.

slowly cooled in the same atmosphere (furnace) to room temperature. The temperature in this process is maintained for larger periods of time and preserves energy for the microstructure to grow. Normalizing on the other hand is accomplished by heating to a critical temperature $T_c$ for an extended period of time and cooling in air to room temperature. It refines grains (decrease the average grain size) and produces a more uniform distribution. It also relieves stresses, increases softness, ductility and toughness.

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Interlamellar structure can be clearly observed at different magnifications. Figure 3, Figure 4 and Figure 5 show scanning electron photomicrographs at 1300 X, 3000 X and 20000 X magnifications respectively. Regions having the alternating light and dark lamellar structure are pearlite; the dark and light layers in the pearlite correspond, respectively, to ferrite and cementite phases. To get mechanical information of the lamellae microstructure, hardness measurements were taken with a Vickers Scale of 0.3 as suggested by the ASTM E384-99 [6] standard.

By using the software ImageJ [7], the average
thickness at a magnification of 20000X was measured in normalized and annealed heat treated specimens. Values shown on Figure 5 and Table 1 reveal that the microstructural and mechanical features are in agreement with the basic kinetic principles of solid-state transformations. The thickness was almost doubled in annealing. Indeed, it affected the hardness as shown on Table 1. Annealing is used to eliminate residual stresses left from a previous process. In metals, this process "restores" a material through the elimination of dislocations from plastic deformation by inducing recrystallization. This process can be divided into three processes: Recovery, recrystallization, and grain growth. Dislocation density decreases significantly during recrystallization effectively eliminating any prior strain hardening. A significant movement of atoms occurs because the annihilation of dislocations density happens, and therefore, the hardness is considerably reduced [8]. Contrary, the higher hardness observed in the normalized condition (fine pearlite) is due to the presence of existing high dislocations density. These dislocations limit the space and free movement of atoms.

Table 1: Average thickness and average hardness of the studied lamellar microstructures.

|                     | Ferrite thickness (µm) | Cementite thickness (µm) | Lamellar thickness (µm) | Hardness HV 0.3 |
|---------------------|------------------------|--------------------------|-------------------------|-----------------|
| Normalizing (fine pearlite) | 0.11 ± 0.0258          | 0.065 ± 0.027            | 0.175                   | 202 ± 18.7      |
| Annealing (coarse pearlite)  | 0.185 ± 0.0717         | 0.109 ± 0.0247           | 0.294                   | 56.42 ± 2.7     |

The annealing and normalizing of AISI 1045 Steel are compared in the present work. As documented on Figure 5 and Table 1, the thickness of ferrite and cementite in the annealed condition is bigger than in the normalized condition. It explains the lower hardness found in the annealed tested specimens. Quantified results demonstrate a strong dependence of flow stress (hardness) on phase type. The results are in agreement with the expected output that the flow resistance is higher for structures composed of finer and homogeneously distributed interlamellar aggregates. That is, hardness increases as grain size decreases when the metal undergoes a fast cooling rate. The microstructure is clearly shown by SEM, and the coarse or fine pearlite can explain those two specimens after different heat treatments.

Conclusions

1. It was verified that the resultant lamellar thickness is smaller in the normalized condition than in the annealed condition. The lamellar thickness in the normalized condition (high cooling rate-fine pearlite) is 0.175 µm, and in the annealed condition (low cooling rate-coarse pearlite) is 0.294 µm.
2. Using the Vickers Hardness Tester, it was determined that the hardness of the AISI 1045 steel in the annealed condition and normalized condition is 56.42 HV and 202 HV, respectively. This corresponds to the fact that hardness increases as grain size decreases when the metal undergoes a fast cooling rate.

3. The results are in agreement with the expected output that the flow resistance (hardness) is higher for structures composed of finer and homogeneously distributed interlamellar aggregates.

4. The effects of microstructural features upon the aggregate local mechanical response are clearly observed.

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