Monolithic integration of embedded III-V lasers on SOI

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Abstract

Silicon photonic integration has gained great success in many application fields owing to the excellent optical device properties and complementary metal-oxide semiconductor (CMOS) compatibility. Realizing monolithic integration of III-V lasers and silicon photonic components on single silicon wafer is recognized as a long-standing obstacle for ultra-dense photonic integration, which can provide considerable economical, energy-efficient and foundry-scalable on-chip light sources, that has not been reported yet. Here, we demonstrate embedded InAs/GaAs quantum dot (QD) lasers directly grown on trenched silicon-on-insulator (SOI) substrate, enabling monolithic integration with butt-coupled silicon waveguides. By utilizing the patterned grating structures inside pre-defined SOI trenches and unique epixial method via hybrid molecular beam epitaxy (MBE), high-performance embedded InAs QD lasers with monolithically out-coupled silicon waveguide are achieved on such template. By resolving the epitaxy and fabrication challenges in such monolithic integrated architecture, embedded III-V lasers on SOI with continuous-wave lasing up to 85 °C are obtained. The maximum output power of 6.8 mW can be measured from the end tip of the butt-coupled silicon waveguides, with estimated coupling efficiency of approximately -6.7 dB. The results presented here provide a scalable and low-cost epixial method for the realization of on-chip light sources directly coupling to the silicon photonic components for future high-density photonic integration.

Introduction

There is fast-growing demand for integrated silicon photonic chips incorporating on-chip lasers, which can lead to power-efficient and densely integrated optical interconnects and high-speed optical communications1–4. Various external lasers coupled silicon integrated chips have been demonstrated on 200 nm silicon-on-insulator (SOI) wafers, however, on-chip lasers remain as the stumbling block for further development of ultra-dense silicon photonic integration5,6. Over the past decade, III-V/Si heterogeneous integration via bonding techniques is academically and commercially recognized as a promising path towards realization of on-chip light sources7–10. With the rapid progresses of silicon-integrated photonics in applications such as artificial intelligence, hyper-scale data centers, high-performance computing, light-detection and ranging (LIDAR) and microwave photonics11, monolithically integrated light sources begin to show their increasing preference as an alternative technical trend for compactness and low-power consumption12.

As monolithic integration of III-V laser and silicon photonic components has always been a much-desired functionality, direct epitaxial growth of III-V quantum-dot (QD) lasers on silicon substrate has been extensively investigated with dramatic progress recent years13–20. Benefiting from the technical development of high-quality III-V material growth on silicon21–25, various silicon-based laser structures have been demonstrated with outstanding performance, including distributed feedback (DFB) lasers26–29, microcavity lasers30–32 and mode-locked lasers33–36. From silicon photonic integration perspectives, all active and passive silicon photonic
components should be on the same SOI platform, therefore, there is a strong urge to develop an SOI-based monolithic laser integration solution to efficiently couple the light into silicon waveguides (WGs). Up to date, there are relatively limited researches focusing on direct growth of SOI-based lasers\textsuperscript{37–40}, and usually the lasers are located on top silicon of SOI with thick III-V buffer layers which are not capable of being coupled with passive waveguides. In case of heterogeneous bonding techniques mentioned above, the coupling from bonded lasers to silicon WGs has been well established by evanescent coupling\textsuperscript{7}. On the other hand, although directly grown III-V lasers have been systematically investigated, the monolithic integration between III-V lasers and silicon photonic components remains absent in the field. In this work, we have demonstrated the first embedded QD lasers on SOI substrate with pre-patterned laser trenches and silicon waveguides, with monolithic coupling to silicon passive waveguides, which offers great potential of achieving fully monolithic silicon photonic integrated chip.

Figure 1a depicts the schematic of monolithically integrated III-V QD lasers edge-coupled with silicon waveguides on SOI substrate. The scanning electron microscope (SEM) image and optical microscope image of fabricated devices are shown in Fig. 1b and c. Starting with an 8-inch SOI wafer, silicon waveguides are prepatterned on top silicon layer of SOI substrate (Fig. 1d). The laser trench is then produced via dry etching process through buried oxide (BOX) layer into bulk silicon substrate for III-V laser growth, as shown in Fig. 1e. Inside laser trench, periodic silicon grating structures are then patterned with the duty cycle of approximately 40% (146 nm slab width with 209 nm gap) as the surface SEM images show in Fig. 1f. The grating structures cover the entire laser trench for high-quality III-V direct epitaxial growth. The zoomed-in SEM image of silicon gratings is displayed in Fig. 1g.

**Results**

**Design and fabrication of SOI template**

Figure 2a presents the fabrication process of the SOI devices. The edge coupler and patterned trench are manufactured on an SOI wafer with a 220 nm thick top Si layer and a 3 μm thick buried SiO\textsubscript{2} layer. The fork-shaped coupler and interconnecting waveguide are defined through E-beam lithography (EBL) process. The resist pattern was fully etched using the ICP-RIE process.
Subsequently, 3 μm thick SiO\(_2\) cladding layer is deposited by plasma-enhanced chemical vapor deposition (PECVD) after removing the e-beam photoresist. The detailed fabrication information of the edge coupler can be found in the Methods. Then, the cladding, top Si, BOX, and 1.5 μm substrate layer are etched to form the laser trench. To note, the pre-defined laser trench area for III-V laser epitaxy is 16 mm × 16 mm for each cell, which exhibit a
III-V fill factor of approximately 16%. Finally, the silicon gratings are fabricated by EBL and ICP-RIE etching for III-V laser growth.

Compared to common inverse taper edge coupler with single tip, edge couplers with multiple taper tips could have higher coupling efficiency and better alignment tolerance for laser butt-coupling scheme due to its elliptical spot-size are more comparable to laser’s mode profile\(^41,42\). Hence, considering both eminent performance and tractable fabrication process, fork-shape edge couplers with double tips are proposed\(^43,44\). Besides the tip width, the fork shape offers an extra design parameter, while the gap between two tips can be also used to expand the mode field (Supplementary Fig. S1).

The structure of the fork-shape coupler is shown in Fig. 2b. The same design methods are followed from our previous work\(^41,42\) with further optimized structures. In order to verify the performance of the edge couplers used in monolithically integrated embedded InAs QD lasers on SOI wafer, we separately fabricate the silicon edge couplers with the same parameters and measured with a single mode fiber (SMF) and InAs QD laser grown on silicon substrate via butt coupling design. The tip width (\(w\)) is set as 100 nm, and the gap (\(g\)) between the two tips is 3.4 \(\mu\)m. The length of the first fork stage (L1) which converts the fiber mode to the slot waveguide mode is 64 \(\mu\)m. The second fork stage, converting the slot waveguide mode to the strip waveguide mode, is 11 \(\mu\)m long. Here, Fig. 2c shows the simulation result of the electric field distribution of this edge coupler. The SMF or the monolithically integrated laser modes, which both exhibit relatively large mode profile, transfer preferentially into the edge coupler facet with large modal overlap, and then adiabatically converts to the strip waveguide mode. The edge coupling loss is mainly determined by the mode similarity, which is also called mode overlap, between the light input and the coupler facet\(^41\). In simulation (Supplementary Section I), the coupling loss between the SMF with mode field diameter (MFD) of 10 \(\mu\)m and the chip facet is 2.41 dB in the O band. In comparison, the coupling loss between InAs QD laser and fork shape coupler is approximately 1.25 dB, which is relatively lower than SMF, which indicate better mode overlap. Notably, the total length of the coupler is only 75 \(\mu\)m, which is superior for the precious integrated photonic chip. In order to achieve optimum spot-size converter, various edge couplers are designed and fabricated for reducing the coupling loss between the monolithically integrated lasers and silicon waveguides (Supplementary Fig. S2).

**Monolithic epitaxial growth and fabrication of III-V lasers on trenched SOI substrate**

Figure 2d shows the entire monolithic process of on-chip lasers, including the embedded growth of III-V gain materials in the grating patterned SOI trench and subsequent laser fabrications. As depicted in Fig. 2d, the pre-patterned silicon waveguides are protected with PECVD SiO\(_2\) top cladding (Supplementary Fig. S3a). The SOI template with pre-defined silicon waveguides is firstly prepared in order to create the laser trench on the bottom silicon of SOI substrate for laser material deposition. To note, as both the horizontal and height alignments to silicon waveguides are critical, the depth of laser trench needs to be precisely designed in line with the active region height of InAs QD laser on silicon. Although, the exact alignment can still be finely adjusted during the epitaxial growth, which normally can be controlled within an accuracy of the nanometer scale. In order to avoid anti-phase domains (APDs), threading dislocations (TDDs) and thermal mismatch-induced thermal cracks\(^45,46\), homoepitaxially formed (111)-faceted Si V-grooves over the top of pre-defined silicon gratings are introduced here to suppress defects generated during hetero-epitaxial growth as shown in step 1 and 2 of Fig. 2d. The SEM images of the structures are presented in Supplementary Fig. S3 (Supplementary Section II). These techniques will be discussed explicitly in the Methods, while the entire epitaxial structures will be described in the following section.

There is one major issue here as observed in step 3 of Fig. 2d, during the hybrid III-V/IV growth process via molecular beam epitaxy, both silicon and III-V materials will also be deposited outside the laser trench in polycrystalline format. In this case, it normally leads to relatively large height contrast in- and outside of the laser trench (Supplementary Fig. S4), which can result in uneven photoresist stacking at the edge of the trench during laser-waveguide alignment process. To solve this problem, \(\text{H}_3\text{PO}_4,\ \text{H}_2\text{O}_2,\ \text{H}_2\text{O}\ (1: 2: 20)\) wet etching solution is selected here to remove excessive polycrystalline III-V materials outside the trench region before the laser fabrication process.

The embedded III-V laser is then processed with one-side cleaved facet (step 5 of Fig. 2d). As previously mentioned, the alignment precision between laser and silicon waveguide will significantly affect coupling efficiency especially in the vertical direction. Meanwhile, the accurate control of alignment in horizontal direction is determined by photolithography during the laser device process. The alignment deviation between the central axis of silicon waveguide and the laser ridge here can be controlled within \(\pm 250\) nm, which is within the tolerance of the unique multipe tips taper design. To ensure precise alignment between the laser ridge and silicon waveguide, the removal process of ploycrystalline silicon and III-V materials above the waveguide has to be treated properly, otherwise, the blurred silicon waveguide structure will lead to reduced accuracy of laser-to-waveguide alignment.

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In addition, the polycrystalline material removing process mentioned above acts also as facet etching for III-V/Si interface at laser output side, as the gap area between laser and silicon waveguide remains un-protected during the removing process. But the wet etching process is insufficient here to create decent facet quality, which shall be repolished during following steps.

The integrated device is finalized by applying high reflection coating to as-cleaved facet at one side, while implementing focused ion beam (FIB) milling process to the other side for the laser output facet as shown in step 6 of Fig. 2d. The ion milling process here aims to create mirror-like facet by removing the polycrystalline material at III-V/IV interface. Although the embedded laser can still lase without FIB treatment, the facet polishing using FIB can further improve the laser performance significantly. The performance differences between wet-etched facets and FIB-etched facets will be compared and discussed in the laser characterization section.

In case of the monolithic epitaxy of embedded III-V lasers, the growth details are described in Methods. The overall schematic of the laser epi-structure on the trenched SOI template is displayed in Fig. 3a. Here, a 10 nm-thick AlAs nucleation layer was first deposited to optimize the GaAs/Si (111) interface and suppress APD formation as marked in Fig. 3a. After approximately 2100 nm-thick III-V buffer layers, which includes InGa(Al)As/GaAs quantum well dislocation filters (DLFs) and GaAs/AlGaAs superlattice (SL) layers, a smooth and APD-free GaAs surface can be achieved on the trenched SOI region as verified by atomic force microscope (AFM) and SEM measurements shown in Fig. 3b. The $5 \times 5 \mu m^2$ AFM image in Fig. 3b shows the root-mean-square (RMS) roughness of only 0.8 nm. Figure 3c shows the surface electron channeling contrast imaging (ECCI) result of the trenched GaAs/SOI template, indicating a $2.6 \times 10^7/cm^2$ surface threading dislocation density (TDD) of the template. The bright-filed cross-sectional transmission electron microscopy (TEM) image at the GaAs/Si(111) interface is also presented in Fig. 3d, which indicates the same defect suppression effect of the homo-epitaxially formed Si(111) sawtooth structures reported previously \cite{32,34}. Additional material characterization details are included in Supplementary Section II.

Based on these high-quality trenched GaAs/SOI templates, the standard InAs/GaAs QD laser diode structures \cite{39,47} are grown as shown in Fig. 3a. The laser structure consists of a 7-layer InAs QD active region, which is sandwiched between the 400 nm n-/p-doped GaAs contact layers and 1400 nm n-/p-doped Al$_{0.4}$Ga$_{0.6}$As cladding layers. Astride each AlGaAs cladding layer, the step-graded Al$_x$Ga$_{1-x}$As ($0.1 < x < 0.4$) transitional layers are deposited in order to increase the current injection efficiency of the device. As Fig. 3a shows, the polycrystalline III-V materials will be formed on the SiO$_2$ cladding layers outside the trenched region, which can be removed by wet-etching process before device fabrication as mentioned above. To clarify the optical gain properties of the InAs QDs on the trenched GaAs/SOI substrate, identical 7 layers of InAs QDs are both grown on the trenched GaAs/SOI (Schematic shown in Supplementary Fig. S5a) and GaAs (001) substrates, respectively,
Characterizations of on-chip integrated lasers

In order to examine the effective coupling efficiency from laser to silicon WGs, here, we select single laser bar directly grown inside SOI trench with both facets cleaved as a reference laser, which is characterized with standard light-current (L-I) measurements (top left of Fig. 5a). Here, the reference laser is produced as an analog to InAs QD laser directly grown on silicon substrate. In case of silicon WG coupled on-chip laser, both L-I curves and optical spectra are collected from the silicon WG side as shown in Fig. 5a. The temperature dependent L-I curves of the embedded laser without silicon WG (so-named as reference laser) are measured in Fig. 5b, which manages to lase up to 95°C in continuous-wave (CW) current operation. The threshold current at room temperature is approximately 50 mA. The maximum output power is 37 mW at injection current of 250 mA.

The silicon WG edge-coupled on-chip laser is then characterized with slightly higher threshold current of 65 mA at room temperature under CW mode, while the maximum operation temperature is reduced by 10°C to 85°C (Fig. 5c). The relatively higher threshold current and lower maximum operation temperature are attributed to increased thermal accumulation inside the laser trench with surrounding BOX layer. The L-I performance of embedded laser with single-side wet etched facet and FIB etched facet are compared in inset of Fig. 5c. With additional FIB facet polishing, the threshold current is reduced from 92 mA to 65 mA, while the out-coupled optical power through silicon WG is also improved from 5.3 mW to 6.8 mW at injection current of 210 mA. Here, both the reference laser and on-chip integrated laser’s cavity length is 3 mm, and the ridge width is 3 μm. Due to relatively large divergence angle of InAs QD laser and slightly height mismatch induced during the gain material growth, the detected output power from silicon WG side is lower than the actual laser power. The overall insertion loss is here estimated to be -7.35 dB (including 0.3 dB from two 90-degree bendings and 0.62 dB transmission loss from the 814 μm long silicon waveguide), which is similar to that -7.6 dB under pulsed operation mode (Supplementary Section III). Furthermore, in order to extract characteristic temperature (T₀) and slope efficiency of both as-cleaved laser and on-chip integrated laser, threshold current and output power at different temperatures are analyzed as shown in Fig. 5d. The as-cleaved laser on trench SOI exhibits T₀ values of 282.4 K and 51 K in temperature ranges from 20 to 40°C and 45 to 95°C. In comparison, on-chip integrated laser (silicon WG output) has slight degradation in T₀ values, that are 108.1 K and 33.7 K in temperature ranges from 20 to 65°C and 70 to 85°C. Moreover, as Fig. 5c shows, the reference laser presents a higher slope efficiency of 0.148 W/A at 20°C, while the silicon WG coupled on-chip laser owns that of 0.025 W/A at 20°C, which are induced from non-optimized edge-coupling efficiency and slightly height mismatch induced misalignment between embedded laser
Fig. 4 Fabricated monolithically integrated InAs QD lasers coupled to silicon waveguides. 

- **a** Tilted SEM image of entire integrated chip.
- **b** SEM image of fabricated narrow ridge laser directly coupled with silicon waveguides using fork-like mode converter.
- **c** SEM image of silicon waveguide mode converter.
- **d** Magnified SEM image of single embedded laser edge-coupled with silicon waveguide.
- **e** Optical microscope of monolithically integrated laser and silicon WG.
- **f** SEM images of the embedded lasers with wet etched and FIB etched facets, respectively.
- **g** White light interferometric image of the finalized devices.
and silicon WG. Fiber collimator is here implemented to collect light from silicon WG side for optical spectral analysis. Current dependent spectral measurements from 70 mA to 160 mA at room temperature are shown in Fig. 5e, with lasing wavelength ranging from 1270 nm to 1283 nm. The spectral evolution of on-chip integrated laser operating in the temperature ranging from 20 °C to 70 °C at fixed injection current of 175 mA is measured in Fig. 5f. The pulsed characterizations of the monolithically integrated laser were also conducted shown in Supplementary Fig. S6. To the best of our knowledge, this is the first demonstration of the InAs/GaAs QD laser that is epitaxially grown on a trenched SOI template with a butt-coupling silicon waveguide.

Discussion

In summary, monolithic integrated III-V lasers on SOI substrate with silicon waveguide output have been realized by directly growing InAs QD lasers inside pre-patterned SOI trenches. Homoepitaxial formation of (111)-faceted Si V-grooves and heteroepitaxial growth of InGaAs/GaAs defect trapping techniques are implemented in this work to achieve high-quality III-V gain materials on SOI. Our results demonstrate that monolithic integration of III-V laser with silicon photonic components will no longer be a design-level hypothesis. Overall, the monolithically integrated lasers can operate over 85 °C with low threshold current of 65 mA at room temperature and silicon WG coupled maximum output power of 6.8 mW. One more step forward, the performance of on-chip integrated InAs QD lasers can be further improved by including advanced silicon spot-size converter with accurate control of laser-waveguide coupling distance during the process. Once the coupling efficiency issue is resolved, many selections of silicon photonic components can all be integrated monolithically on a single wafer, such as modulators, wavelength de-multiplexers and photodetectors, just to name a few. Meanwhile, QD laser growth on a 300 mm patterned (001) silicon wafer with butt-coupled configuration has also been recently demonstrated, which leads to a great prospect for achieving dense on-chip integration. The next major step towards functional integration shall be including single longitudinal mode operation with high side mode suppression ratio (SMSR), which shall require implementing
surface gratings or on-chip distributed bragg reflectors. Overall, we believe that this monolithic integration techniques of on-chip lasers would offer a promising approach towards high-density and large-scale silicon photonic integration, especially in the application fields such as on-chip optical interconnect and integrated optical ranging.

Materials and methods

Fabrication of silicon edge couplers

The fork shape coupler and interconnecting waveguide are defined through a single E-beam lithography (EBL) process with Vistec EBPG-5200+ Electron-beam lithography system. The patterns on 400 nm thick AR-P 6200.09 photoresist are fully etched using the SPTS ICP Deep Silicon Etching System, the flow of C4F8 and SF6 are 45 sccm and 20 sccm, respectively. Subsequently, after removing the E-beam photoresist, a 3 μm thick SiO2 cladding layer is deposited by plasma-enhanced chemical vapor deposition (PECVD) process with OXFORD Plasmalab System 100, at 300 °C temperature. The cladding, top Si and BOX layers are etching with NMC ICP Reactive Ion Etching System, the flow of CHF3 and Ar are 80 sccm and 35 sccm, respectively. Afterwards, 1.5 μm depth of the substrate Si layer are etched with SPTS ICP Deep Silicon Etching System to form a trench. Finally, the gratings are fabricated on the substrate in the trench by EBL and ICP etching for the laser growth.

Material growth

The epitaxial materials were prepared by our dual chamber solid-source MBE system (hybrid III-V/IV) to allow in-situ transfer between III-V and IV chamber without exposing to atmosphere. The SOI substrates are first diced into 3.2×3.2 cm² pieces, which are suitable for the 4-inch MBE system, and cleaned with standard RCA process to remove the surface contaminants. Before loading into MBE chamber, the SOI template was dipped into diluted HF solution (HF: H₂O = 1: 10) strictly within 10 seconds to remove the native oxide layer in the "U"-shape grating-patterned trench region, while avoiding horizontal wet-etching of the BOX layers surrounding the trench region and surface etching of SiO₂ top cladding layer. The epitaxial growth is initiated with 420 nm thick homoepitaxial growth of silicon buffer layer at substrate temperature of 600 °C with a 1 Å/s growth rate in Group IV MBE chamber, to construct uniform sawtooth structures with Si (111) Miller facets in the trench SOI region. The substrate was then in-situ transferred into III-V growth chamber for the following growth of III-V buffer layers and InAs/GaAs QD laser structure. To note, although the geometry of the trench SOI substrates with passive silicon WGs is much different from the initial SOI substrate, the disturbance in substrate growth temperature is still negligible in III-V epitaxial growth process. After in-situ transferred into GaAs MBE chamber, the substrate was heated to 360 °C for growth of 10 nm AlAs and 30 nm GaAs nucleation layers. Then, 560 nm GaAs buffer layer was grown at 540 °C with a 1 Å/s growth rate to flatten the unevenness caused by the sawtooth structures. One repeat of 10 times 10 nm In₀.₁₃Ga₀.₈₇As/10 nm GaAs quantum wells and two repeats of 10 times 10 nm In₀.₁₅Al₀.₈₅As/10 nm GaAs quantum wells were grown at 440 °C as dislocation filters, which are separated by 200 nm GaAs strain-relaxed buffer layers. After depositing 3 periods of Al₀.₄Ga₀.₆As/GaAs SLs, the trenched GaAs/SOI template with smooth surface and low TDD in trenched region was achieved. The detailed schematic of III-V buffer layers grown on the trenched SOI was shown in Supplementary Fig. S5a.

On the trenched GaAs/SOI template, 400 nm thick heavily n-doped GaAs contact layer was first grown at 540 °C, following by 300 nm Al₀.₄Ga₀.₆As transitional layers, 1400 nm Al₀.₄Ga₀.₆As cladding layer, and 200 nm step-graded upper AlₓGa₁₋ₓAs transitional layers, which were all n-doped. The active region includes 7 layers of InAs QDs, which was sandwiched by 50 nm and 11 nm unintentionally doped (UID) GaAs layers as shown in Fig. 3a. Each of the InAs QD layers consists of 8.1 Å InAs QD, 4 nm In₁.₄₅Ga₀.₅₅As capping layer and 39 nm GaAs spacer layer. The InAs QD and capping layers are grown at an optimized temperature of 420 °C, which is 20 °C less than that used on GaAs/Si (001) and GaAs (001) substrates. Above the QD active region, symmetric upper AlGaAs structures were grown at 560 °C but with p-type doping. At last, a 400 nm heavily p-doped GaAs contact layer was deposited at 540 °C with a 1 Å/s growth rate. Overall, the total thickness of deposited III-V materials are 7.05 μm inside SOI trenches.

Fabrication of on-chip integrated lasers

The integrated lasers were fabricated from embedded growth of III-V gain materials on SOI described above by using the standard lithography and dry etching techniques. For the definition of 3 μm wide ridges, DWL66 + Laser Direct Writing system with SPR 220 photoresist is used here to allow flexibility of laser-waveguide alignment. The ridges are dry-etched with Plasmalab System 100 ICP180 system (BCl₃: Cl₂: Ar₂ = 10: 6: 4) for 2.2 μm etching depth. P-type and N-type metal contacts are deposited as Ti/Pt/Ti/Au and Ni/Ge/Au/Ti/Au by using PEVA-600E electron beam evaporation system and DZ-300 thermal evaporation sedimentary system, respectively. Additional 2.3 μm thick inter-metal contact were deposited under three separate processes with different deposition angles (-45°, 45°, 0°), which exhibit thickness of 20/700 nm, 20/700 nm, and 20/900 nm, correspondingly. After the completion of laser process, the substrate was thinned to 70 μm for single-facet cleaving. The length of
the laser cavity is 3 mm. The laser’s facet that is away from the waveguide was cleaved to form laser cavity. HR coating was also applied to this facet. The HR coating consists of 8 pairs of SiO2 (145 nm) and Nb2O5 (228 nm) superlattice. The reflection of the HR coating is measured as 95%.

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Author contributions
W.Q.W. and D.H. performed the direct epitaxial growth of III-V laser inside SOI trenches. B.Y. fabricated the integrated laser device. A.H. performed the design and fabrication of SOI template. J.J.Z. provided overall technical leadership and fabrication of SOI template. B.Y., J.Z.H. and Z.H.W. performed the trench fabrication. T.W., H.G., Y.S. and J.J.Z. provided overall technical leadership and fabrication of SOI template. B.Y. fabricated the integrated laser device. A.H. performed the design and fabrication of SOI template. J.J.Z. provided overall technical leadership and fabrication of SOI template. B.Y., J.Z.H. and Z.H.W. performed the trench fabrication. T.W., H.G., Y.S. and J.J.Z. provided overall technical leadership and fabrication of SOI template.

Data availability
The data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

Conflict of interest
The authors declare no competing interests.

Supplementary information
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