UV-Enhanced Humidity Sensing of Chitosan–SnO$_2$ Hybrid Nanowires

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Abstract: The surface of SnO$_2$ nanowires was functionalized by chitosan for the development of room-temperature conductometric humidity sensors. SnO$_2$ nanowires were synthesized by the seed-mediated physical-vapor-deposition (PVD) method. Chitosan layers were deposited on top of the SnO$_2$ nanowires by spin coating. Surface morphology, crystal structure, and optical properties of the synthesized hybrid nanostructure were investigated by scanning electron microscope, grazing incidence X-ray diffraction, and UV–Vis absorption measurements. During electrical conductivity measurements, the hybrid nanostructure showed unusual behavior towards various relative humidity (RH) concentrations (25%, 50%, 75%), under UV-light irradiation, and in dark conditions. The highest sensor responses were recorded towards an RH level of 75%, resulting in 1.1 in the dark and 2.5 in a UV-irradiated chamber. A novel conduction mechanism of hybrid nanowires is discussed in detail by comparing the sensing performances of chitosan film, SnO$_2$ nanowires, and chitosan@SnO$_2$ hybrid nanostructures.

Keywords: chitosan; SnO$_2$; hybrid; nanostructures; room temperature; humidity sensor

1. Introduction

Organic conductive polymers are used in many varied applications [1–3]. Due to their extended $\pi$-conjugation of the polymer chain by alternating double bond–single bond structure, partial oxidation causes variations in electrical conduction with the formation of charges on their chains [4]. In this way, some, including polyaniline (PAni), polypyrrole (PPy), and polythiophene (Pth), form chemoresistive surfaces with their special molecular structures [5–7]. They alter their electrical conductance when they are exposed to some gases. They have many advantages: ease of construction, optimum performance at room temperature, low power consumption, rapid and reversible adsorption/desorption kinetics, low toxicity, and high dependency on analyte concentration compared to inorganic materials [8].

Chitosan is a well-known polymer formed by repeating units of 1,4-linked 2-deoxy-2-amine glucose. It has noteworthy properties such as film-forming ability, biocompatibility, biodegradability, hydrophilicity, and, importantly, non-toxicity. The main feature that makes it favorable is the use of chitin, one of the most abundant polysaccharides in nature, as source material [9]. Many studies are currently working on chitosan-derivative products because of their sustainable, eco-friendly, and economic benefits.

It is an insulating polymer, but protonated amine groups in its chains make it conductive and sensitive to polar gas vapors. Therefore, metals, carbon nanotubes, metal oxides, and conductive...
polymers mixed with chitosan layers have been tested as gas-sensitive materials. For instance, Mironenko et al. studied the changes in optical waveguides made of Ag nanoparticles mixed chitosan and Au nanoparticles mixed chitosan composites in the presence of H₂S gas [10]. Abu-Hani et al. fabricated flexible H₂S sensing films by adding glycerol to chitosan polymers [11]. Kumar et al. discovered the chemo-electrical behavior of carbon nanotube (CNT) blended chitosan layers towards water, methanol, and toluene vapors [12]. Li et al. synthesized PANi–chitosan nanocomposites for hydrogen sensors [13]. Ayad et al. revealed dimethylamine sensing properties of chitosan blended polyaniline nanofibers [14]. Similarly, Wang et al. decorated chitosan nanofibers with polyethyleneimine to detect formaldehyde [15]. In most of these studies, chitosan has been used as a film-forming material, not as an active sensing material. In addition to these, chitosan was used in more complex composites for gas sensing [16,17]. Molla-Abbasi and Ghaffarian suggested the use of chitosan-decorated carbon nanotube surfaces for polar chemical vapors [18]. They discovered improved humidity sensing abilities of chitosan@CNT heterostructures related with the chitosan surfaces. Dai et al. also verified these observations by presenting the humidity sensing properties of chitosan-decorated ZnO–SWCNT (single-walled carbon nanotube) structures [19]. Xu et al. increased the humidity sensing abilities of bare ZnO nanorods by four times by using a chitosan layer on top of ZnO nanorods [20].

To date, the research has focused on the usage of chitosan to get an advantage from its hydrophilic character. A hybrid structure of chitosan and SnO₂ nanowires (NWs) was proposed for a conductometric humidity sensor using a similar approach, as tin oxide (SnO₂) has been the most attractive sensing material with its tailor-made semiconducting properties [21]. Its charge carriers have a high concentration and high mobility at elevated temperatures [21,22]. Considering the instability of chitosan at high temperature, UV light irradiation was used to benefit from the semiconducting properties of SnO₂ NWs.

2. Materials and Methods

2.1. Synthesis of SnO₂ Nanowires

A seed-layer-assisted vapor–liquid–solid (VLS) technique was used for the synthesis of SnO₂ NWs. Pre-cleaned alumina substrates (3 × 3 mm²) were coated with an Au seed layer by sputtering at a 6.6 × 10⁻⁴ mbar working pressure and 50 W RF magnetron power (Ar plasma) for 5 s. The VLS growth was carried out in a physical-vapor-deposition (PVD) furnace at 100 mbar (Ar flow of 100 sccm) for 2 min. The SnO₂ powder was placed in the middle of the furnace and heated up to 1370 °C, while the substrates were placed at a small distance from SnO₂ dust at a temperature of 860 °C.

2.2. Synthesis of Chitosan@SnO₂ Hybrid Nanowires

A chitosan solution was prepared by dissolving 0.3 g low-molecular-weight chitosan powder (Sigma-Aldrich, Darmstadt, Germany, CAS no: 9012-76-4) in acetic acid aqueous solution (1.5%). Then, the chitosan solution was deposited on top of SnO₂ nanowires by spin coating. A 50 µL aliquot of the prepared solution was dripped onto rotating samples at 6000 rpm. The spinning time lasted 45 s after two drops. Then, samples were taken into a clean place to allow for drying of the layer for 48 h.

2.3. Sample Characterization

The formation of SnO₂ nanowires and the deposition of chitosan layers were monitored by a field-emission scanning electron microscope (FE-SEM, Leo 1525 Gemini model; Carl Zeiss AG, Oberkochen, Germany). Surface images of chitosan@SnO₂ nanowires were taken under small potential ranges (1–5 kV) and in tilted positions (45° and 75°) to avoid damaging the chitosan layers. The crystalline properties of SnO₂ were demonstrated by X-ray diffractometry (Empyrean; PANalytical, Almelo, Netherlands) measurements in glancing angle mode (with an incident angle of 1.5°). A Cu-LFF (λ = 1.5406 Å tube was used for X-ray emissions and operated at 40 kV/40 mA. The spectra were
recorded by using a Xe detector in the range of 20–60°. Optical properties of SnO\textsubscript{2} and chitosan@SnO\textsubscript{2} nanowires were determined by a UV-vis spectrometer (UV-2600, Shimadzu, Kyoto, Japan).

### 2.4. Conductometric Humidity Measurements

Transducers with SnO\textsubscript{2} and chitosan@SnO\textsubscript{2} nanowires were equipped with contacts by magnetron sputtering to carry out electrical measurements. The TiW and Pt contact pads were deposited at 75 W DC power (Ar plasma) and 6.6 × 10\textsuperscript{-3} mbar working pressure for 3 min respectively in order to get adhesion. Then, a Pt interdigitated transducer (IDT) was deposited under the same pressure and DC power conditions for 20 min (≈1 µm thickness). The sensors, integrated into TO-8 cases, were placed in a climatic test chamber for humidity measurements. An integrated hybrid sensor is shown below in Figure 1.

![Image of a chitosan@SnO\textsubscript{2} hybrid nanowire (NW) sensor.](image)

The conductometric changes were monitored by picoammeter during humid air injections and dry air flows. The tested relative humidity (RH) concentrations were 25%, 50%, 75%, 50%, and 25% respectively. The measurement was repeated in the same conditions under UV light irradiation. The UV light source had a ~365 nm wavelength (\(\lambda_{\text{max}}\)); however, a 350 nm Dichroic filter (T = 30%) was used during the measurement to control the irradiation power of 0.5 µW/mm\textsuperscript{2}. The response values were calculated according to conduction type (n- or p-) of chitosan and hybrid nanowires using the following formulae:

\[
\text{Response (R)} = \frac{G_{\text{air}} - G_{\text{gas}}}{G_{\text{gas}}} \quad (n-\text{type}) \quad (1)
\]

\[
\text{Response (R)} = \frac{G_{\text{gas}} - G_{\text{air}}}{G_{\text{air}}} \quad (p-\text{type}) \quad (2)
\]

### 3. Results and Discussion

#### 3.1. Material Characterizations

The formation of SnO\textsubscript{2} nanowires followed the VLS mechanism. The Au nano-seed layer became liquid at 860 °C @100 mbar. The SnO\textsubscript{2} powder evaporated and the Ar flow acted as a gas carrier to transport SnO\textsubscript{2} vapors from the source to the substrates. The formation of SnO\textsubscript{2} nanowires was triggered by the condensation of the SnO\textsubscript{2} vapors on top of the Au seed layers and by the ensuing segregation at the bottom of the droplet. The fabricated SnO\textsubscript{2} nanowires were homogenous through the substrate surface (Figure 2). They were 2–4 µm long and 50–80 nm thick.
The GI-XRD profile of SnO\(_2\) nanowires is shown in Figure 3. The characteristic peaks of SnO\(_2\) nanowires belonged to (110), (101), (211), and (220) crystalline planes located at 26.8°, 33.3°, 52.2°, and 54.8°, respectively. The rutile crystalline phase of SnO\(_2\) nanowires was consistent with the standard JCPDS 41-1445 pattern. The polycrystalline alumina substrate peaks were marked with "*" in the GI-XRD spectrum. The small quantity of Au peaks, around 38° and 44°, displayed the low presence of Au traces. It is a common mechanism of VLS that the catalyst (Au) comes up on the top of nanowires (also visible in Figure 2b).

Despite the low viscosity of the chitosan solution, the chitosan layers were successfully deposited by spin coating. At 6000 rpm rotation rates, the drop of chitosan spread through the surface of SnO\(_2\) NWs. The surface morphologies of the chitosan@SnO\(_2\) NWs hybrid (Figure 4) were investigated by SEM at 45° (a) and 75° (b) tilted positions (Figure 4). Chitosan layers could be seen as blurred sheets connecting the nanowires. They did not cover the surface of nanowires completely but filled the empty spaces between nanowires.
The conductance value increased exponentially at the level of 25% RH. Additionally, it decreased exponentially after a rapid increase at 50% and 75% RH levels. The maximum response value of bare SnO$_2$ NWs was measured at around $1 \times 10^{-7}$ S, drawn in red. SnO$_2$ NWs did not exhibit any significant changes towards water vapor injections in either measurement. However, the baseline conductance value of chitosan@SnO$_2$ hybrid NWs increased approximately by one order of magnitude, from $8 \times 10^{-7}$ S to $1.5 \times 10^{-5}$ S, under UV light. A similar change was observed with the conductance measurement of ZnO thin film coated bacterial cellulose membranes in previous work [27]. In the dark chamber, humidity changes led to variations in the response profiles of chitosan@SnO$_2$. The conductance value increased exponentially at the level of 25% RH. Additionally, it decreased exponentially after a rapid increase at 50% and 75% RH levels. The maximum response values were recorded as 1.1 in the dark, and 2.5 under UV-light-irradiated measurements both towards 75% RH (at 20 °C). The detection limit of the hybrid sensor was determined as 5% RH for the UV-irradiated

Figure 5. Tauc’s plot and UV–Vis absorbance spectra.

3.2. Humidity Sensing

Figure 6 reports the dynamic sensor responses of both the SnO$_2$ and chitosan@SnO$_2$ hybrid NWs towards humidity changes in the dark (a) and UV-light-irradiated (b) chambers. The baseline value of bare SnO$_2$ NWs was measured at around $1 \times 10^{-4}$ S, drawn in red. SnO$_2$ NWs did not exhibit any significant changes towards water vapor injections in either measurement. However, the baseline conductance value of chitosan@SnO$_2$ hybrid NWs increased approximately by one order of magnitude, from $8 \times 10^{-7}$ S to $1.5 \times 10^{-5}$ S, under UV light. A similar change was observed with the conductance measurement of ZnO thin film coated bacterial cellulose membranes in previous work [27]. In the dark chamber, humidity changes led to variations in the response profiles of chitosan@SnO$_2$. The conductance value increased exponentially at the level of 25% RH. Additionally, it decreased exponentially after a rapid increase at 50% and 75% RH levels. The maximum response values were recorded as 1.1 in the dark, and 2.5 under UV-light-irradiated measurements both towards 75% RH (at 20 °C). The detection limit of the hybrid sensor was determined as 5% RH for the UV-irradiated
measurement by using the calibration curve (dotted line) as seen in Figure 7 and by considering 0.5 as the minimum significant response value.

Figure 6. Dynamic sensor responses of bare SnO$_2$ and chitosan@SnO$_2$ hybrid NWs in the dark (a) and under UV light irradiation (b).

Figure 7. Response values of hybrid nanowires under UV light vs. measured relative humidity (RH) levels in logarithmic scales.

Figure 8a reports the humidity-sensing capabilities of the chitosan film on an alumina substrate. Despite its lower surface area and its lower conductivity, the chitosan film showed sensor signals proportional to humid air concentrations. The conductance value increased by the injections of moist air, similar to the dark measurement profile of chitosan@SnO$_2$ hybrid nanowires. As can be easily deduced by comparing the humidity-sensing performances of bare SnO$_2$ and bare chitosan film, chitosan was the main actor in the humidity sensing of hybrid NWs. In previous studies, the sensing mechanism of chitosan was explained by the proton-conduction mechanism [28,29]. The adsorbed water molecules on the surface cause neutralization of protonated amine groups in chitosan chains. The primary electrochemical process on the chitosan surface has been described by the following equation [29]:

$$\text{R…NH}_3^+ + \text{H}_2\text{O} \rightarrow \text{R…NH}_2 + \text{H}^+$$

(3)
was dominated by protonic conduction, as given before in Equation (3). The protonic conduction threshold RH value between 25–50% RH to activate valance electrons of SnO\textsubscript{2} in Figure 9b. The attracted valance electrons neutralized the agglomerated electrons of SnO\textsubscript{2} on a CNT–chitosan surface [30]. Similarly, the protonic conduction of chitosan might create an electrical potential at the chitosan–SnO\textsubscript{2} interface. The protons participated in the surface reactions and sensing mechanism.

The humidity-sensing mechanism of hybrid chitosan@SnO\textsubscript{2} nanowires [31] is shown in Figure 9c. At low levels of humidity, water molecules adsorbed onto the SnO\textsubscript{2} surface and reduced the protonic conduction, as given below in Equation (4):

$$e^- + H^+ \rightarrow \frac{1}{2} H_2$$

(4)

The dual conduction was not observed in 25% RH. This observation indicates that there is a threshold RH value between 25–50% RH to activate valance electrons of SnO\textsubscript{2}.

**Figure 8.** (a) Humidity measurement profile of the chitosan film. (b) Zoomed conductance profile of chitosan@SnO\textsubscript{2} nanowires hybrid in dark condition towards 75% RH (at 20 °C) injection.

The dynamic measurement data of the hybrid sensor were plotted on a small scale in Figure 8b. In I and II regions, indicated by dashed lines in Figure 8b, the hybrid sensor had dual conduction behavior under a constant humid air concentration (75% RH). In region I, the conductance value increased suddenly after water vapor injection. The electrical conduction of the hybrid nanostructures was dominated by protonic conduction, as given before in Equation (3). The protonic conduction reduced the resistance of the chitosan layer rapidly and caused a decrease in the equivalent resistance of the hybrid system. In region II, the conductance started to decrease after reaching a peak point during the constant flow of water vapor. It was an interesting observation since the bare SnO\textsubscript{2} sample did not show any significant response to water vapor injections. The only possible explanation could be that the valance electrons of SnO\textsubscript{2} participated in the surface reactions and sensing mechanism. Rana et al. highlighted a change in the dielectric environment after the absorption of vapor molecules on a CNT–chitosan surface [30]. Similarly, the protonic conduction of chitosan might create an electrical potential at the chitosan–SnO\textsubscript{2} interface. The H\textsuperscript{+} agglomeration on the surface attracted the valance electrons of SnO\textsubscript{2} and caused an increase in the equivalent resistance of the hybrid system, as shown in Figure 9b. The attracted valance electrons neutralized the agglomerated H\textsuperscript{+} ions and reduced the protonic conduction, as given below in Equation (4) [31]:

$$e^- + H^+ \rightarrow \frac{1}{2} H_2$$

(4)

**Figure 9.** The humidity-sensing mechanism of hybrid chitosan@SnO\textsubscript{2} nanowires (a) in a low level of humidity, (b) in a high level of humidity, and (c) under UV irradiation.
3.3. UV Effect on Sensing

There was no significant effect of UV light on the dynamic measurement profile of bare SnO$_2$ NWs, as seen in Figure 6b. Besides, the measurement profile of the chitosan@SnO$_2$ hybrid NWs was reversed from p-type to n-type under UV illumination. Additionally, the response values of the hybrid sample were increased compared to dark measurements. Even though SnO$_2$ was the active material for UV absorption, proven by the optical analysis above, hybrid nanocomposite sensors showed better performance compared to bare SnO$_2$. The UV light irradiation generated electron–hole pairs in the SnO$_2$ surface. In the literature, some studies proposed that SnO$_2$ and its heterostructure showed better NO$_2$ detection under UV light because generated photoelectrons captured more O$_2$ molecules on the surface, resulting in band bending [32–34]. Similarly, in the present case, the UV light may have bent the potential energy barrier between chitosan and SnO$_2$ NWs. Thus, valence electrons could take place in the surface reactions. This proposed mechanism explains why hybrid sensors reversed the conduction type under UV irradiation and why there were no dual conduction behaviors at levels of 50% and 75% RH as observed in the dark measurement profiles.

Table 1 summarizes the response values of the chitosan@SnO$_2$ hybrid NWs and some other chemoresistor-type chitosan-based humidity detectors. A direct comparison of response values cannot be fair for the performance evaluation since the reported studies applied different test procedures and defined response parameters with different methodologies. For example, in two of the studies, response values were fitted by using the Langmuir–Henry–Clustering (LHC) model to correlate gas sensing with properties of CNT–chitosan and polyvinyl alcohol (PVA)–chitosan–CNT blended films [12,18]. Thus, they used nitrogen as a carrier gas during their measurements. One study has reported the best response value with chitosan films deposited by spin coating on a quartz substrate [29]. However, the baseline conductance value of chitosan film was around 0.1 nS, which is challenging for real applications. After considering differences between previous studies, our hybrid sensors exhibited satisfactory and more practical performance.

Table 1. Comparison table of chitosan-based conductometric humidity sensors.

| Material               | Transducer  | RH Conc.         | Response | Reference |
|------------------------|-------------|------------------|----------|-----------|
| 2% CNT-Chitosan        | Chemoresistor | 1% Vapor Conc.   | 1        | [12]      |
| PVA/Chitosan/CNT       | Chemoresistor | 10 % Vapor       | 0.05     | [18]      |
| Chitosan/ZnO/SWCNT     | Chemoresistor | 75%              | 0.75     | [19]      |
| Chitosan               | Chemoresistor | α% RH            | 4.2 × α  | [29]      |
| Chitosan@SnO$_2$ Hybrid Nanowires | Chemoresistor | 75% RH (dark)    | 1.1      | This work |
|                        |             | 75% RH (UV irradiation) | 2.5     | This work |

4. Conclusions

SnO$_2$ NWs were combined with chitosan layers to take advantage of the hydrophilic property of chitosan for a room-temperature humidity detector. The study aimed to increase the surface area of the chitosan layer and to get biocompatible sensing layers and transducer material. Chitosan polymer was deposited by a spin-coating method on top of SnO$_2$ NWs. The humidity sensing of the chitosan@SnO$_2$ NWs hybrid showed better sensing properties than both chitosan thin films and bare SnO$_2$ NWs. The dual conduction behavior of hybrid NWs was explained with a new mechanism combining the protonic conduction and the effect of the chitosan@SnO$_2$ NWs hybrid. The surface reactions, possible chemical interactions, and effects of UV irradiation were discussed, and a sensing mechanism was proposed. Chitosan@SnO$_2$ hybrid NWs exhibited better sensing performances under UV light irradiation.
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