Dielectric and electrical properties of ZnSb$_2$O$_4$ ceramics

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Received: 06 January 2013, Revised: 22 March 2013 and Accepted: 16 April 2013

ABSTRACT

The polycrystalline sample of ZnSb$_2$O$_4$ was prepared by a high-temperature solid-state reaction technique. Preliminary X-ray diffraction (XRD) studies of powder sample of ZnSb$_2$O$_4$ showed the formation of single-phase compound at room temperature. The surface morphology of the pellet sample of ZnSb$_2$O$_4$ was recorded at room temperature using a scanning electron microscope (SEM). Detailed studies of dielectric properties ($\varepsilon'$, $\tan \delta$) and impedance parameters of the material provide an insight into the electrical properties and understanding of types of relaxation process occurred in the material. Temperature variation of dc conductivity shows that this compound exhibits negative temperature coefficient of resistance (NTCR) and frequency dependence of ac conductivity suggests that the material obeys Jonscher’s universal power law. Copyright © 2014 VBRI press.

Keywords: Ceramics; XRD; electrical conductivity; impedance spectroscopy.

Introduction

Nowadays, a large number of oxides of different structural families are available in different forms (ceramics, thin films and single crystals) for scientific and industrial applications. Some of them, such as BaTiO$_3$ [1], PbZrTiO$_3$ [2], ZnO [3] etc are widely used for variety of electronic and other applications. Out of many non-linear devices, varistors are found voltage dependent, which has electrical behavior similar to that of back-to-back Zener diodes [3, 4]. Because of non-linear I-V characteristic and energy absorption capability, ZnO-based varistor modified with small amount of Bi$_2$O$_3$, Sb$_2$O$_3$, Mn$_2$O$_3$, Al$_2$O$_3$, Cr$_2$O$_3$etc have extensively been studied [5-9]. Not much information on structural, capacitive and resistive behavior of a new composition of ZnO (with above oxides in equal ratio) is available. Therefore, we have attempted to synthesize and characterize ZnSb$_2$O$_4$. This material has extensively been studied in order to (i) enhance its electrical properties and (ii) to solve some inherent problems so that the material can be used as a non-linear device. In this paper we report some important features of dielectric and electrical properties.
Experimental

Materials
Polycrystalline samples of ZnSb$_2$O$_4$ were prepared from pure oxides, ZnO (99.9 %, M/s. LobaChemie. Pvt. Ltd., India) and Sb$_2$O$_3$ (99.9 %, M/s. LobaChemie. Pvt. Ltd., India) in a suitable stoichiometry.

Synthesis
A conventional high-temperature solid-state reaction technique was used to synthesize the compounds. The ingredients were thoroughly mixed and ground in dry condition for 1h, and in wet atmosphere (methanol) for 1h in an agate mortar and pestle to get homogenous mixture of the material. Then the mixed powders of the compound were calcined at ~1150°C in alumina crucibles for ~12 h in air atmosphere. The process of grinding and calcinations were repeated to ensure the formation of the compounds. The quality and formation of the compound was verified by an X-ray diffraction (XRD) technique. The fine and homogeneous powder of the above compound was pressed into cylindrical pellets of 10 mm diameter and 1-2 mm thickness under a uniaxial pressure of 5x10$^6$N/m$^2$ using a hydraulic press. Polyvinyl alcohol (PVA) was used as a binder to reduce the brittleness of the pellets. The pellets were then sintered for ~10h at ~1150°C in air atmosphere. The binder was burnt out during the sintering. Crystal structure and microstructure of the prepared compound was studied by an X-ray diffraction technique (XRD) and scanning electron microscopy (SEM) respectively. The sintered pellets were polished with fine emery paper and then coated with high-purity conducting silver paste. The pellets were dried at 150°C for about 4 h.

Characterization
A preliminary structural analysis was carried out using X-ray diffraction (XRD) technique by using an X-ray powder diffractometer (Rigaku Mini flex, Japan) in order to confirm the formation of single phase compound. The XRD pattern of the calcined powder was recorded at room temperature with CuK$_\alpha$ ($\lambda = 1.5405$ Å) in a wide range of Bragg angles 20 ($20^\circ \leq 2\theta \leq 65^\circ$) at a scan rate of 3°/minute. The microstructure or texture of the pellet surface was recorded by a scanning electron microscope (SEM).

Electrical measurements were carried out on silvered-coated pellet samples. The capacitance in parallel mode ($C_p$), dissipation factor (tan δ), impedance (Z) and phase angle (Φ) of ZnSb$_2$O$_4$ were obtained as a function of frequency (1kHz-1MHz) at different temperatures(30-500°C) using a computer-controlled LCR/impedance meter (PSM 1735, N4L) with a laboratory-made furnace. A chromel-alumel thermo-couple and digital panel-voltmeter was used to measure the temperature. The I–V characteristics were studied on the sintered-pellet of the compound, and the data were recorded using a computer-controlled sensitive electrometer (Keithley, model 6517B).

Results and discussion
Structure/microstructure
Fig. 1 shows sharp and single XRD peaks, which are different from that of ingredients, suggest the formation of a single-phase new compound [10]. A good agreement between observed and calculated interplanar spacing (d) was found in tetragonal crystal system at room temperature. The selected lattice parameters (a and c) of a tetragonal unit cell were refined using least-squares refinement sub-routine of POWD [11]. All of the diffraction peaks were indexed to those of the zinc antimoniate (ZnSb$_2$O$_4$) with a spinel structure of lattice constants:$a = 4.6408$ Å, $c = 18.6398$Å. The surface morphology of the pellet sample (Fig1 (inset)) shows small size grains homogeneously-distributed throughout the surface of the sample.

**Fig. 1.** X-ray diffraction pattern and (inset) SEM micrograph of ZnSb$_2$O$_4$ at room temperature.

Dielectric study
The nature of variation of tanδ with temperature follows the similar pattern as of $\varepsilon_r$ as shown in Fig. 2. The dielectric anomaly is assumed to be related to the phase transition. The sharp increase in tanδ at higher temperatures may be

**Fig. 2.** Variation of $\varepsilon_r$ and tan δ with temperature at selected frequencies.
due to scattering of thermally activated charge carriers and some defects in the sample [12].

Fig. 3. Variation of $Z'$ and $Z''$ at selected temperatures (Impedance fitting).

Table 1. Comparison of dc conductivity, $\omega$ and $n$ at different temperatures.

| $T$ (°C) | $\sigma_{dc}$ (ohm$^{-1}$m$^{-1}$) | $\omega$     | $n$     |
|----------|---------------------------------|-------------|--------|
| 400      | 0.000003                        | 4.45E-7     | 0.6116 |
| 425      | 0.000004                        | 2.0984E-7   | 0.57209|
| 450      | 0.000005                        | 4.45E-7     | 0.53675|
| 475      | 0.000007                        | 3.3219E-7   | 0.59028|
| 500      | 0.000009                        | 7.918E-6    | 0.36968|

Fig. 4. Variation of $Z'$ & $Z''$ with frequency at selected temperatures.

Electrical properties: Complex impedance spectroscopy

Complex impedance spectroscopy (CIS) was used to study the electrical properties of the compound. The frequency dependent properties of materials can be described via the complex dielectric permittivity ($\varepsilon^*$), complex impedance ($Z^*$), complex admittance ($Y^*$), and dielectric loss (tan $\delta$). These parameters are related to each other by the following relations: $Z^* = Z' - jZ'' = 1/j\omega C_0$; $\varepsilon^* = j\omega C_0 Z^*$; $Y^* = Y'$ + $jY''$ and $\tan \delta = \varepsilon''/\varepsilon' = Z''/Z' = Y''/Y'$. Where $\omega = 2\pi f$ is the angular frequency, $C_0$ is the geometrical capacitance, $j = \sqrt{-1}$.

Fig. 4 shows the temperature dependence of complex impedance spectra (Nyquist plot) of the said compound over selected frequencies. The ZIMP WIN version of software was used to analyze the data. In an ideal case (Debye-like response) an equivalent circuit has parallel combination of CQR and CR where Q is known as constant phase element (CPE), R is the resistance and C is the capacitance. Value of $n$ is between zero and one (for an ideal capacitor $n=1$ and for an ideal resistor $n=0$) [13].

Using the fitted curves, the values of bulk resistance ($R_b$) and bulk capacitance ($C_b$) at different temperatures were calculated and compared in Table 1. The decrease in the value of $R_b$ on increasing temperature indicates the existence of NTCR behavior in the material as generally observed in semiconductors [14]. No grain boundary and interfacial effect were observed. Depressed semi-circles indicate the presence of non-Debye type of relaxation [15, 16]. The intercept of each semi-circle on real $Z'$-axis gives the contributions of bulk in the resistance/impedance. Fig. 4 shows the variation of real and imaginary parts of impedance (i.e., $Z'$ and $Z''$) with frequency at different temperature. The value of $Z'$ decreases with rise in temperature and frequency. It suggests reduction in bulk resistance. It is also observed that the value of $Z'$ decreases till a certain fixed frequency up to 100 kHz, and attain a constant value above this frequency implying the possible release of space charge [17]. This indicates the absence of frequency relaxation process in the material [13]. The decrease in $Z''_{max}$ and the broadening of $Z''_{max}$ peak with increase of temperature suggests the occurrence of temperature dependence of relaxation phenomenon in the material due to the presence of immobile charges at low temperatures and defects and vacancies at higher temperatures [18 - 20].

Fig. 5. Variation of dc conductivity with temperature.

DC conductivity

Fig. 5 shows the temperature dependence of dc conductivity of the material. It is observed that $\sigma_{dc}$ increases with increase in temperature which supports the
NTCR behavior of the sample. The nature of the plot follows the Arrhenius relation

$$\sigma_{dc} = \sigma_0 \exp \left( - \frac{E_a}{k_B T} \right)$$ [19].

The calculated activation energy ($E_a$) of the sample in the temperature range of 27 – 76 °C is 0.24 eV, whereas it is 0.452 eV in the temperature range of 125 – 176 °C. The variation of current density ($J$) with applied dc electric field ($E$) at 30 °C and 150 °C is shown in Fig. 6.

![Fig. 6. Variation J with E selected temperatures.](image)

AC conductivity

The ac electrical conductivity ($\sigma_{ac}$) was calculated using the dielectric data in an empirical relation: $\sigma_{ac} = \epsilon_0 \epsilon_r \tan \delta$, where $\epsilon_0$ is permittivity in free space and $\omega$ is angular frequency. In order to have better understanding of conduction mechanism in the material, Jonscher’s universal power law [21] is used:

$$\sigma_T(\omega) = \sigma(0) + \sigma_1(\omega) = \sigma_0 + \alpha \omega^n$$

where $\sigma(0)$ is the frequency independent term giving dc conductivity and $\sigma_1(\omega)$ is the purely dispersive component of ac conductivity having a characteristic of power law behavior in terms of frequency ($\omega$). The exponent $n$ can have a value between zeros to one. Fig. 7 shows the variation of ac conductivity of the material as a function of frequency at different temperatures with non-linear fitting.

![Fig. 7. Variation of ac conductivity with frequency.](image)

| $T$(°C) | $R_0$(kΩ) | $C_b$(pF) |
|---------|-----------|-----------|
| 400°C   | 1.85E+006 | 9.84E-011 |
| 425°C   | 1.36E+006 | 9.94E-011 |
| 450°C   | 7.75E+005 | 9.79E-011 |
| 475°C   | 4.54E+005 | 9.71E-011 |
| 500°C   | 2.82E+005 | 8.64E-011 |

The conductivity plots reveal that the curves exhibit low frequency dispersion phenomena obeying the Jonscher’s power law equation. According to Jonscher [19], the origin of the frequency dependence of conductivity lies in the relaxation phenomena arising due to mobile charge carriers. From the graphs it is obvious that $\sigma_{ac}$ increases with rise in frequency but it is nearly independent at low frequency. The extrapolation of this part towards lower frequency side gives $\sigma_{dc}$. In the high frequency region the curves approach each other. From non-linear fitting it is found that that motion of charge carriers is translational because of small value of $n$ ($< 1$) [22]. The values of dc conductivity and $n$ have been compared in Table 2.

**Table 2** Impedance-fitting data of ZnSb$_2$O$_4$

**Conclusion**

Finally, it is concluded that the titled compound ZnSb$_2$O$_4$ has tetragonal structure at room temperature. This compound shows dielectric anomaly. The contribution to the impedance comes from bulk only where the effect of grain boundary and interface is insignificant. This compound also exhibits negative temperature coefficient of resistance, which indicate the semi-conducting character of the material.

**Acknowledgements**

The authors are grateful to Dr. Maiti (CRF of IIT, Kharagpur), Dr. Dilip Mishra of IMMT, Bhubaneswar, Mr. R. Padhee and Samita Pattanaik of ITER for their help in experimental work and discussion.

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