Fabrication, Characterization, and Modeling of an Aluminum Oxide-Gate Ion-Sensitive Field-Effect Transistor-Based pH Sensor

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Abstract
The ion-sensitive field-effect transistor (ISFET) is a popular technology utilized for pH sensing applications. In this work, we have presented the fabrication, characterization, and electrochemical modeling of an aluminum oxide (Al₂O₃)-gate ISFET-based pH sensor. The sensor is fabricated using well-established metal–oxide–semiconductor (MOS) unit processes with five steps of photolithography, and the sensing film is patterned using the lift-off process. The Al₂O₃ sensing film is deposited over the gate area using pulsed-DC magnetron-assisted reactive sputtering technique in order to improve the sensor performance. The material characterization of sensing film has been done using x-ray diffraction, field-emission scanning electron microscopy, energy-dispersive spectroscopy, and x-ray photoelectron spectroscopy techniques. The sensor has been packaged using thick-film technology and encapsulated by a dam-and-fill approach. The packaged device has been tested in various pH buffer solutions, and a sensitivity of nearly 42.1 mV/pH has been achieved. A simulation program with integrated circuit emphasis (SPICE) macromodel of the Al₂O₃-gate ISFET is empirically derived from the experimental results, and the extracted electrochemical parameters have been reported. The drift and hysteresis characteristics of the Al₂O₃-gate ISFET were also studied, and the obtained drift rates for different pH buffer solutions of 4, 7, and 10 are 0.136 μA/min, 0.124 μA/min, and 0.108 μA/min, respectively. A hysteresis of nearly 5.806 μA has been obtained. The developed sensor has high sensitivity along with low drift and hysteresis.

Keywords ISFET · pH sensor · sputtering · XRD · XPS · SPICE macromodel

Introduction
In recent years, there has been a surge in the demand of handheld point-of-care (POC) electrochemical devices for environmental, biomedical, and food quality monitoring applications. These devices are useful for chemical as well as biosensing applications, owing to several advantages such as small size, fast response and accuracy.¹,² Broadly, a chemical/biochemical sensor captures physicochemical signals from an analyte and converts them to a measurable signal, comprising light emission, colorimetric readout, or potential/current modulation.² There are various techniques for signal transduction using optical and non-optical techniques, such as surface plasmon resonance, quartz crystal microbalance, micro-cantilever, and electrochemical techniques. An ion-sensitive field-effect transistor (ISFET) is an electrochemical-based potentiometric technique which has potential to provide real-time and selective detection of chemical and biological analytes, along with the compatibility with the well-established complementary metal–oxide–semiconductor (CMOS) fabrication process.³,⁴ It is a solid-state device, which makes it rugged for the measurement of ion activities in analytes with the aid of a reference electrode,⁵ and it has been widely used for pH measurements. pH is one of the most important parameters in agricultural, environmental, biomedical, and potable water studies.⁶,⁷

The ISFET has attracted much interest due to the feasibility of facile integration with readout circuits.⁸ It is an alternative to the conventional glass membrane electrode-based
pH sensors, where it enables measurements in ultrasmall volumes with fast response. It is used for chemical as well as biochemical sensing applications by suitable surface functionalization, and recently, it has been demonstrated for detecting severe acute respiratory syndrome coronavirus 2 (SARS-CoV-2) virus, which causes COVID-19. For pH sensing applications, various sensing film materials have been studied in the past, such as aluminum nitride (AlN), silicon dioxide (SiO_2), silicon nitride (Si_3N_4), tantalum pentoxide (Ta_2O_5), etc. Several physical and chemical deposition techniques are used to grow the thin films, such as pulsed laser deposition (PLD), magnetron sputtering, chemical vapor deposition (CVD), sol–gel technique, and atomic layer deposition (ALD). Sputtering is a popular cost-effective process for depositing thin films. Sputtering is carried out by utilizing different variants of power supplies, which include radio frequency (RF), direct current (DC), and pulsed-DC power supply. In order to deposit insulating thin films, the DC reactive sputtering process is not feasible due to the formation of an insulating film on the surface of the target, which results in an operating potential shift due to poisoning of the target, which is undesirable. Although these problems have been resolved by using RF power supplies, they provide low deposition rates and require tedious process optimization to obtain desired network matching required for an efficient sputtering process. Pulsed-DC sputtering is a popular cost-effective technique which addresses these issues for depositing thin films. Pulsed-DC sputtering facilitates easier process optimization because it does not require costly and complex matching networks required for the RF sputtering process, which makes it more widely adopted in the industrial production units worldwide. Moreover, when sputtering is assisted by a magnetron, the sputtering efficiency increases, leading to higher film growth rate, and the obtained film quality is better than the films grown under similar process conditions, viz., operating pressure and substrate temperature, in comparison to conventional sputtering techniques, as described by the structure zone model.

In this work, we have deposited aluminum oxide (Al_2O_3) thin film using pulsed-DC magnetron-assisted reactive sputtering technique. Aluminum oxide is amphoteric in nature and has a wide bandgap, with excellent mechanical, optical, and electrical properties. It is a robust material which is chemically stable and wear resistant, which enables it to be used for widespread applications, such as abrasive materials, electrical insulators, cutting tools, etc. Moreover, it is a biocompatible material, which allows it to be used for dentistry and orthopedic applications. Al_2O_3 material has also been studied for pH sensing applications, where it has been demonstrated to have very good hysteresis characteristics and low long-term drift, which makes it a promising material to be used as a sensing film of an ISFET. We have fabricated an Al_2O_3-gate ISFET pH sensor using the self-aligned process, with the sensing film patterned using a lift-off process. The experimental results are empirically validated through Simulation Program with Integrated Circuit Emphasis (SPICE) macromodeling, and the extracted electrochemical parameters have been reported.

We obtained a high sensitivity and low drift using Al_2O_3 as a sensing film of the ISFET. To the best of our knowledge, this is the first time Al_2O_3 sensing film grown using cost-effective pulsed-DC magnetron-assisted reactive sputtering approach has been demonstrated for ISFET pH sensor applications. In the past, Al_2O_3 thin films have been deposited using CVD, ALD processes in order to be used as sensing films for ISFET pH sensor applications. Both CVD and ALD processes have inherent disadvantages in comparison to sputtering processes, although the quality of deposited films are good. CVD processes involve high temperatures for the deposition of Al_2O_3 thin films. The high-temperature processing is not acceptable in the presence of metallic layers such as aluminum in the device structure, which makes the CVD incompatible with the developed self-aligned process in this work, where the sensing film is deposited and patterned as the last step by using the lift-off process. Moreover, CVD requires specialized precursors, which increases the process complexity. In addition, the precursors and the by-products are toxic and pyrophoric, which is undesirable for commercial production of ISFET pH sensors. ALD processes for deposition of thin films are very costly due to the requirement of high-grade precursors which are toxic and pyrophoric. They also provide a very slow growth rate, which is its major limitation. Moreover, it involves tedious process optimization and is commercially not viable due to low throughput. In contrast, sputtering is a simpler, low-temperature, cost-effective, and environmentally friendly process for thin-film deposition, suitable for commercial production of ISFET pH sensors.

The paper is further organized as follows: Section “ISFET Principles and Theory” discusses the theory of ISFET sensor operation and the site binding model. Section “Material and Methods” describes the process optimization of sensing film deposition and fabrication process of the sensor. Results and discussion have been presented in the section “Results and Discussion,” which consists of material and device characterization results. Finally, conclusions are presented in the section “Conclusions.”

**ISFET Principles and Theory**

ISFET is a FET-based chemical sensor popularly used for pH sensing applications. Unlike a metal–oxide–semiconductor field-effect transistor (MOSFET), the gate electrode is absent in an ISFET, and the gate region is left exposed during device
packaging to allow interaction of the gate oxide with analyte solutions. The sensor is immersed in the analyte along with a reference electrode, where the solution comes in contact with the exposed gate region as shown in Fig. 1. A reference potential is applied to the solution with the aid of a reference electrode. When immersed in analyte, due to the formation of charges at the insulator–electrolyte interface, the channel conductivity is modulated, leading to accumulation or depletion of charges near the oxide–silicon interface, which causes a shift in the threshold voltage \( V_{TH} \) of the transistor. The threshold voltage of an ISFET depends upon the electrochemical parameters and surface potential at the solid–electrolyte interfaces; i.e. between the analyte and the gate oxide on one side, and the analyte and the reference electrode on the other side.

The expression for current between drain and source of the ISFET in the unsaturated region is given by:

\[
I_{DS} = K_n \left( V_{GS} - V_{TH} \right) - \frac{V_{DS}}{2} \]

where \( I_{DS} \) is the drain current; \( K_n \) is the transconductance parameter; \( V_{GS} \) is the gate source voltage; \( V_{TH} \) is the threshold voltage; and \( V_{DS} \) is the drain source voltage. The threshold voltage expression for the ISFET is given as:

\[
V_{TH} = E_{ref} - \Psi_0 + X^{sol} - \frac{\phi_{Si}}{q} - \frac{Q_{SS} + Q_{OX} + Q_B}{C_{OX}} + 2\psi_F
\]

where \( E_{ref} \) is the reference electrode potential; \( \Psi_0 \) is the surface potential at the insulator–electrolyte interface; \( X^{sol} \) is the surface dipole potential of the solution; \( \phi_{Si} \) is the work function of Si (substrate); \( q \) is the elementary charge; \( Q_{SS} \) is the accumulated charge at the \( \text{O}_2/\text{Si} \) interface; \( Q_{OX} \) is the accumulated charge in the oxide; \( Q_B \) is the depletion charge in the silicon; \( C_{OX} \) is the oxide capacity per unit area; and \( \psi_F \) is the Fermi potential. Here, \( \Psi_0 \) is the only parameter which depends on the pH of the solution. The surface potential of the electrolyte/insulator interface can be calculated using the site-binding model, and which was further elucidated by Bousse et al. and Fung et al. to characterize ISFETs with various gate insulators. Figure 2 shows a schematic of the surface sites of \( \text{Al}_x\text{O}_y \) sensing film, where the surface of sensing film has amphoteric sites capable of accepting or donating protons, and where Al represents the metal. The resultant surface charge density depends upon the pH of the analyte and the electrochemical parameters of the sensing film, consisting of dissociation constants and the intrinsic buffer capacity. Thus, the surface sites have three different forms which are negatively charged (AlO\(^-\)), positively charged (AlOH\(^2+\)), and neutral sites (AlOH).

**SPICE Macromodel Formulation**

The ISFET device is considered as a combination of two stages, i.e. an electrochemical stage, which contains the electrolyte/insulator interface, and an electronic stage, which consists of a MOSFET. Let \( \sigma_{diff} \), \( \sigma_0 \), and \( \sigma_J \) represent...
the charge densities in the diffuse layer, at the electrolyte/insulator interface, and in the semiconductor, respectively. By using the condition of charge neutrality of the structure:

$$\sigma_{\text{diff}} + \sigma_0 + \sigma_s = 0 \quad (3)$$

Assuming $\sigma_s$ to be much smaller than $\sigma_{\text{diff}}$ and $\sigma_0$, and considering it to be constant with pH, 58 Eq. 3 reduces to:

$$\sigma_{\text{diff}} + \sigma_0 = 0 \quad (4)$$

Using the electrical double-layer theory and site-binding model,57,59–61 we obtain

$$\sigma_{\text{diff}} = \sqrt{8\varepsilon \varepsilon_0 kT} \sinh\left(\frac{z\psi_{\text{sd}}}{2kT}\right) \quad (5)$$

where $\psi_{\text{sd}}$ is the potential across the diffuse layer; $\varepsilon_0$ and $\varepsilon$ are the permittivity of the electrolyte and free space, respectively; $T$ is the temperature (kelvin); $k$ is the Boltzmann constant; $C_b$ is the concentration of bulk solution (no./lit); $z$ is the valence of the ion; and $Q$ is the elementary charge.

$$\sigma_o = qN_s\left(\frac{a_{H^+}^2 - K_aK_b}{K_aK_b + K_b(a_{H^+} + a_{H^+}^2)}\right) \quad (6)$$

where $N_s$ is the surface site density; $K_a$ and $K_b$ are the intrinsic dissociation constants for deprotonation and protonation reactions, respectively; and $a_{H^+}$ is the proton activity near the electrolyte/insulator interface. Figure 3 shows a potential diagram for the ISFET, depicting the potential across the diffuse layer ($\psi_{\text{sd}}$) and the surface potential ($\psi_o$).

Equation 4 can also be written as:

$$\sigma_{\text{diff}} = -\sigma_0 = -C_{\text{Diff}} \cdot \psi_0 \quad (7)$$

where $C_{\text{Diff}}$ represents the double-layer capacitance, which is a series combination of diffuse layer capacitance and Stern capacitance. As stated by the Gouy–Chapman–Stern–Graham model, an electrical double layer is formed in the electrolyte near the electrolyte/insulator interface due to the presence of surface charges, consisting of an inner Helmholtz plane (IHP) and outer Helmholtz plane (OHP), which are the locus of centers of adsorbed ions and solvated ions, respectively; and a diffuse layer is present away from the OHP towards the bulk of the electrolyte.9,17,63

This leads to the formation of Stern capacitance ($C_{\text{Helm}}$) and diffuse layer capacitance ($C_{\text{Gouy}}$) in the electrolyte,58,62 whose expressions are given as follows:

$$C_{\text{Helm}} = \frac{\varepsilon_{\text{Helm}} \varepsilon_0 d_{\text{OHP}} + \varepsilon_{\text{OHP}} d_{\text{HHP}}}{2VL} \quad (8)$$

$$C_{\text{Gouy}} = \frac{\sqrt{8\varepsilon_0 \varepsilon kT}}{2V_F} \quad (9)$$

where $\varepsilon_{\text{Helm}}$ and $\varepsilon_{\text{OHP}}$ are the relative permittivity of the OHP and IHP, respectively; $d_{\text{OHP}}$ and $d_{\text{HHP}}$ are the locus of centers of planes of the OHP and IHP, respectively; $k$ is the Boltzmann constant; $W$ and $L$ are the gate width and gate length of the ISFET sensor, respectively; $\varepsilon_0$ is the dielectric constant of water; $\varepsilon$ is the permittivity of free space; $V_F$ is the thermal voltage; and $C_b$ is the concentration of the electrolyte. Thus, the expression for $C_{\text{Diff}}$ is given by:

$$C_{\text{Diff}} = \frac{C_{\text{Helm}}C_{\text{Gouy}}}{C_{\text{Helm}} + C_{\text{Gouy}}} \quad (10)$$

By using 6, 7, and 10, we obtain the potential of the electrolyte/insulator interface as:

$$\psi_0 = qN_s\left[\frac{a_{H^+}^2 - K_aK_b}{K_aK_b + K_b(a_{H^+} + a_{H^+}^2)}\right] \quad (11)$$

The SPICE macromodel is simulated using the electrical equivalent model of coupled electrochemical and electronic stages as discussed in the section “SPICE Macromodel Formulation,”17 where a new subcircuit block is defined. The subcircuit block consists of five terminals, viz., source, drain, bulk, an independent pH input terminal, and a reference electrode.43 The electrochemical parameters of the ISFET are empirically extracted from the experimental results by parametric sweep, and the closely fitted results are reported. The electrical device parameters are extracted using MOSFET test structures fabricated alongside ISFET devices on the same wafer and validated using process simulations performed in Silvaco® AthenaTM. The device parameters of fabricated ISFET devices are listed in Table I.

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Fig. 3 Schematic of ISFET showing the potential diagram (adapted with permission from Ref. 62, copyright 2000, Elsevier).
Table I ISFET device parameter values.

| Parameters                          | Value         |
|-------------------------------------|---------------|
| Width                               | 500 μm        |
| Length                              | 20 μm         |
| Gate oxide thickness                | 50 nm         |
| Aluminum oxide thickness            | 100 nm        |
| Sheet resistance (source and drain) | 1.74 Ω/□      |
| Substrate doping                    | 2E+15 cm⁻³    |
| Junction depth                      | 1.2 μm        |
| Lateral diffusion                   | 0.96 μm       |

**Material and Methods**

ISFET-based pH sensors have the advantage of being batch fabricated using well-established MOS unit processes, which significantly reduces the production cost of these sensors. Moreover, the fabrication process and device design is simple, which reduces elaborate unit process optimization requirements. In this work, Al₂O₃-gate ISFET-based pH sensors were fabricated using the self-aligned process. The self-aligned process involves five levels of lithography, where the final step of lithography step involves a lift-off process to pattern the Al₂O₃ thin film. The sensing film was deposited using pulsed-DC magnetron-assisted reactive sputtering technique. The sensing film material was characterized using X-ray diffraction (XRD), field-emission scanning electron microscopy (FESEM), energy-dispersive spectroscopy (EDS), and x-ray photoelectron spectroscopy (XPS) techniques.

**Process Optimization for the Deposition of Sensing Film**

Pulsed-DC magnetron-assisted reactive sputtering technique has been used for the deposition of Al₂O₃ thin film. The target potential switches in the frequency range of 10–350 kHz from negative potential to either positive or ground potential in a pulsed-DC power supply.³⁰ This is helpful in avoiding target poisoning. During the negative potential cycle, a thin layer of dielectric is formed over the target, which reduces the sputtering efficiency of bombarding ions. Moreover, the ions begin to charge the dielectric, which may lead to dielectric breakdown over the target surface, leading to melting and ejection of droplets of target material, which degrades the quality of thin film deposited over the substrate. However, when the potential switches to a positive value, the secondary electrons in the plasma get attracted towards the target and neutralize the charged dielectric, and during the next negative cycle, the ions bombard the target and remove the dielectric layer formed on target surface. Thus, the pulsed-DC power supply enables target cleaning during the reverse pulse duration. The film quality, deposition rate, and stoichiometry of the deposited film are influenced by several process parameters, which include operating power, operating frequency, °C target-to-substrate distance, °C operating frequency, and reverse time. The process parameters are experimentally optimized to obtain the desired stoichiometry, film quality, and thickness. It has been reported that a higher operating frequency, longer reverse time, and higher deposition temperature is desirable to obtain good quality of thin-film growth.

The deposition process for Al₂O₃ sensing film has been developed using the TFSP-840® sputtering system (VST, Israel) using a bipolar pulsed-DC power source. A number of iterations were carried out for optimizing the Al₂O₃ film deposition process. A number of combinations of pulsed-DC power in the range of 200 W to 500 W were used with varying flow rates of argon and oxygen gases for the deposition of the Al₂O₃ thin film. We used a 99.99% pure aluminum target for the thin-film deposition. Initially, a base pressure of 1.25x10⁻⁷ kPa was achieved in the process chamber of the sputtering system. After achieving the base pressure, the substrate was maintained at 100°C during the deposition process. A process pressure of nearly 0.82 Pa was maintained by controlling the gas flow of argon (Ar) and oxygen (O₂) in the ratio 4:1. On achieving stable chamber pressure, a 350-W pulsed-DC power was applied to the target with an operating frequency of 100 kHz and a reverse time of 3 μs. On the stabilization of applied power, the shutter in front of the aluminum target was opened to begin the deposition process. The substrate holder was rotated at 20 RPM at a distance of 60 mm the target to achieve uniform thin-film deposition. The deposition process was carried out for 10 min to achieve nearly 100-nm thickness of Al₂O₃ sensing film. Table II lists the optimized parameters used for the deposition of the sensing film.

**ISFET Device Fabrication**

The fabrication process of an ISFET is similar to a MOSFET; however, the fundamental difference is the absence of gate electrode in an ISFET. In addition to this, a sensing film is stacked on top of the gate oxide to enhance the sensing performance. Moreover, the encapsulation and packaging process of an ISFET is different from a MOSFET because the gate region having sensing film must be exposed to the electrolyte to allow the formation of an electrolyte/insulator interface in order to measure the pH sensitivity.

Figure 4 shows the fabrication process of an ISFET sensor. We use a p-type < 100 > silicon wafer with resistivity of 1–10 Ω-cm for device fabrication (Fig. 4a). Firstly, using thermal oxidation method, a 1-μm-thick oxide layer was grown on both sides of the wafer using a wet oxidation process (Fig. 4b). Next, the source and drain regions...
were patterned using photolithography, followed by etching of oxide using buffered hydrofluoric acid (BHF) (Fig. 4c). Phosphorus diffusion and the drive-in process were carried out in the source and drain regions while masking the other regions with field oxide, and a sheet resistivity of nearly 1.74 Ω/□ (Fig. 4d) was obtained. Next, after performing second-level lithography, the oxide in the gate region and backside were etched using BHF (Fig. 4e). In order to grow gate oxide, dry oxidation was carried out to obtain a SiO₂ layer thickness of around 50 nm (Fig. 4f). Further, contact windows were patterned using third-level lithography, and the oxide was etched using BHF solution (Fig. 4g). Aluminum was further deposited using the DC sputtering process on both sides of the wafer for making electrical contacts (Fig. 4h). The aluminum was patterned using fourth-level lithography, and the metal was etched away using aluminum etchant (Transene, MA, USA) (Fig. 4i), followed by sintering in forming gas. Next, the final level of lithography was carried out to form a lift-off pattern for the sensing film (Fig. 4j). Al₂O₃ sensing film was deposited using the optimized process recipe as discussed in “Process Optimization for the Deposition of Sensing Film” (Fig. 4k), and the lift-off process was carried out to pattern the sensing film (Fig. 4l). Finally, the wafer was diced, and individual devices were packaged in order to use the sensor for pH sensing applications.

### Packaging of Sensor

The device was packaged using thick-film alumina technology. Graffy HYDE® software was used to make the layout design used to fabricate screen masks for the screen-printing process. The base was fabricated using screen printing on alumina substrates of size 2 × 2 in., where the conductor tracks and lead attachment pads were fabricated using Pd-Ag paste, followed by firing and dicing. The wire bonding pads were fabricated using Au paste, and the conductor tracks were protected using alumina substrate. The lead attachment was done using a reflow soldering process, and the devices were die-bonded using conductive adhesive compound. This was followed by the wire bonding process using 1-mil-diameter Au wire and conductive adhesive compound, and curing was done for 10 min at 120°C. Finally, an alumina cap was placed over the conductor tracks to isolate the electrical contact leads. The device was encapsulated using the dam-and-fill process, which is used to create a dam around the device using a high-Viscosity potting compound, which prevents flow of epoxy into the gate area, surrounded by a low-Viscosity compound (Fig. 4m). The curing time was experimentally optimized.

### Results and Discussion

#### Material Characterization

The sensing film was characterized by various material characterization techniques. For XRD characterization, grazing
angle x-ray diffraction (GAXRD) was used. As shown in Fig. 5, the peak of the XRD spectrum of the sensing film was obtained at 45°, and the corresponding plane of the Al₂O₃ thin film was (400). The surface morphology of the sensing film was characterized using FESEM. The obtained film was uniform, as shown in Fig. 6.

EDS analysis was also carried out to determine the stoichiometry of the deposited sensing film. The elemental composition is listed in Table III which includes O₂, Al, and Si elements. Figure 7 shows the obtained EDS spectra of Al₂O₃ thin film.

XPS is a popular technique used to study the chemical composition of surfaces. Here, we analyze the oxidation state of the sensing film. The XPS spectral lines of Al₂O₃ thin film are shown in Fig. 8c. Figure 8a shows the spectrum of the Al 2p energy region of the aluminum oxide. The peak is observed around 75.6 eV, which matches closely with the earlier reported data, where the binding energy of the Al 2p for Al₂O₃ thin film is around 74.4 eV to 75.8 eV. The typical XPS spectrum of the O 1s is shown in Fig. 8b. Two oxygen peaks are observed using the peak deconvolution method. The deconvoluted peaks are observed at 532.3 eV and 532.8 eV. The peak at 532.3 eV originated from the Al–O bond due to the adsorbed oxygen. The other peak at around 532.8 eV originated from the Al–OH hydroxyl group due to exposure of the film to the atmosphere (moisture) before the XPS characterization.

**Al₂O₃-gate ISFET Characterization**

The packaged Al₂O₃-gate ISFET was immersed in pH buffer solutions (Merck) of pH values 4, 7, and 10 along with a Ag/AgCl reference electrode (Metrohm®). The drain terminal was supplied a fixed voltage (1 V), while the source and bulk terminals were grounded. The reference voltage was swept from −1 V to +1 V. A Keysight® B2902A source measure unit (SMU) was used for measuring the electrical characteristics of the ISFET sensor, and the electrical schematic of the test setup is shown in Fig. 9. We utilized a 1-kΩ current-limiting resistor to prevent high current flow through the device. The obtained transfer characteristics of the device are shown in Fig. 10. We obtained a sensitivity of nearly 42.1 mv/pH using the following relationship:

\[
\text{Sensitivity} = \frac{\Delta V_{\text{Ref}}}{\Delta \rho H}
\]  

The sensitivity was calculated at a constant current of \( I_{DS} = 0.55 \) mA using the constant-current voltage method. The sensor has a nonlinear behavior over the pH range 4–10.
Figure 11 shows the output characteristics of the \( \text{Al}_2\text{O}_3 \)-gate ISFET for different pH buffer solutions (pH = 4, 7, and 10). A variation in the drain current is observed for different pH values, indicating the dependence of channel conductance on the pH of buffer solutions. For a given pH buffer solution, the drain current is strongly dependent upon the applied reference voltage. Since the fabricated sensor is a depletion-mode ISFET, we obtained a considerable drain current for negatively biased \( V_{\text{Ref}} \), while the current magnitude increased at higher \( V_{\text{Ref}} \) due to the formation of higher inversion charge near the semiconductor–insulator interface. The drain current saturates for higher \( V_{\text{DS}} \) values, indicating operation of the ISFET in the saturation region. As the pH of buffer solutions are varied from pH = 4 (acidic) to pH = 10 (basic), we obtained a lower drain current for the corresponding \( V_{\text{Ref}} \) values due to the reduction in the net

Fig. 8 XPS characterization of \( \text{Al}_2\text{O}_3 \) thin film: (a) Al 2p; (b) O 1s; (c) XPS spectra for aluminum oxide showing core levels of Al 2p, C 1s, and O 1s.

Fig. 9 Electrical schematic of test setup for ISFET pH sensor characterization, where R, D, S, and B represent the reference electrode, drain, source, and bulk terminals, respectively.

Fig. 10 Transfer characteristics of \( \text{Al}_2\text{O}_3 \)-gate ISFET pH sensor.
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positive charges formed at the insulator–electrolyte interface, on account of reduced [H+] ion concentration, as we move from acidic to basic solution. The charges formed at the insulator–electrolyte interface are a function of the pH of the solution, as described in Eq. 6, which influences the channel conductance due to the generation of surface potential as derived in Eq. 11. From Eq. 11, it can be deduced that a net positive charge in acidic solution will reduce the threshold voltage of the ISFET, as explained in Eq. 2, leading to higher drain current for acidic solutions in comparison to basic solutions.

The obtained sensor performance is better than an AlN-gate ISFET, where the reported sensitivity is 33 mV/pH,17 a ZnO nanostructure-based pH sensor, which has a sensitivity of 31.81 mV/pH,76 a SiO2-gate ISFET where a sensitivity of 35 mV/pH was obtained,77 and a Si3N4 sensing film-based ISFET sensor, where a sensitivity of 40.35 mV/pH was achieved.78 Table IV provides a performance comparison of pH sensors using sensing films grown by the sputtering process.

### Table IV: Sensing performance comparison of Al2O3 thin film with other sensing films grown by sputtering techniques.

| Sensing film | Deposition technique | Sensitivity | Reference |
|--------------|----------------------|-------------|-----------|
| AlN          | Pulsed-DC reactive sputtering | 33 mV/pH | 17        |
| ZnO          | RF sputtering         | 42.45 mV/pH| 79        |
| CeO2         | RF sputtering         | 48.62 mV/pH| 80        |
| ZnO          | RF sputtering         | 31.81 mV/pH| 76        |
| TiO2         | RF sputtering         | 37.73 mV/pH| 81        |
| Dy2O3        | Reactive sputtering   | 48.6 mV/pH  | 82        |
| Al2O3        | Pulsed-DC reactive sputtering | 42.1 mV/pH | This work|

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### Al2O3-gate ISFET Macromodeling

The SPICE macromodel of the Al2O3-gate ISFET is empirically derived from the experimental results. The electrochemical parameters are swept to fit the obtained experimental results, and the optimized parameters are listed in Table V. As shown in Fig. 12, the empirical SPICE macromodel matches closely with the experimental transfer characteristics of the ISFET. The nonlinear characteristics of the sensor are attributed to the effect of various electrochemical parameters, such as surface site density and dissociation constants for surface sites, which influence the overall surface charge on the sensing film. The macromodel was simulated using HSPICE® and did not face any convergence issues.

The surface site density (N_s), deprotonation (K_a), and protonation (K_b) dissociation constants are material-dependent parameters. In the past, surface site densities for different sensing films, such as SiO2, Si3N4, Al2O3, AlN, and Ta2O5 have been reported,17,62,83 where the films have been deposited using various processes, including CVD, ALD, sputtering, and thermal oxidation techniques. The characteristics of the sensing film are strongly

![Fig. 11](image_url) Output characteristics of the Al2O3-gate ISFET in different pH buffer solutions: (a) pH 4; (b) pH 7; (c) pH 10.
The obtained silanol surface site density is higher than in order to achieve a close fit with the experimental data. We obtained the electrochemical parameters for the sensing film using a parametric sweep of the parameters for the sensing film using a parametric sweep of the process temperature, which significantly influences the material’s electrochemical parameters. As discussed in "Process Optimization for the Deposition of Sensing Film," we utilized a low-temperature sputtering process for depositing Al2O3 sensing film, which is desirable for commercial production of sensors. We obtained the electrochemical parameters for the sensing film using a parametric sweep in order to achieve a close fit with the experimental data. The obtained silanol surface site density is higher than AlN sensing film, which is expected due to the higher sensitivity obtained using Al2O3 sensing film. The sensitivity of the ISFET pH sensor is strongly dependent upon the surface site density. As discussed by Van Hal et al., the higher density leads to increased intrinsic buffer capacity, which determines the capability of sensing film to rapidly respond to changes in the surface pH. The intrinsic buffer capacity is strongly dependent upon the surface site density, which is given by Eq. 15.

\[ \beta_{\text{int}} = 2.3a_{H^+}N_s\left(\frac{K_aK_b + K_ba_{H^+} + a_{H^+}^2}{(K_aK_b + K_ba_{H^+} + a_{H^+}^2)^2}\right) \]  

(15)

The dimensionless sensitivity parameter \( \alpha \) varies from 0 to 1, depending upon the value of \( \beta_{\text{int}} \) and \( C_{\text{Diff}} \). The double-layer capacitance \( C_{\text{Diff}} \) is nearly constant for different sensing films, as reported by van Hal et al., and thereby the magnitude of intrinsic buffer capacity \( \beta_{\text{int}} \) plays a critical role in determining the sensitivity of the ISFET and significantly affects the surface potential \( \psi_o \), which is responsible for the shift in threshold voltage of the sensor. In comparison to SiO2 sensing film, the reported \( \beta_{\text{int}} \) of Al2O3 film is much higher, which is responsible for higher sensitivity of the Al2O3-gate ISFET pH sensor. Moreover, a small difference between the surface dissociation constants maximizes the intrinsic buffer capacity, as seen in Eq. 15, and a higher value of \( \beta_{\text{int}} \) is desirable. As stated by Fung et al., the dissociation constants \( K_a \) and \( K_b \) for Al2O3 film are closer to each other in comparison to SiO2 sensing film, which aids in obtaining higher sensitivity for the Al2O3-gate ISFET pH sensor.

### Sensor Drift and Hysteresis Characterization

Figure 13 shows the drift characteristics of the sensor in pH 4, 7, and 10 buffer solutions. In order to study the drift behavior, both the drain voltage and the reference voltage were fixed at 1 V, while the source and bulk terminals were grounded, and the drain current was recorded using a Keysight B2902A dual-channel SMU. The measurements were recorded for 3600 s for each of the pH buffer solutions. The drift rates were calculated by dividing the total drift by the elapsed time. The accuracy of the sensor is limited by drift and hysteresis values.

The primary causes for sensor drift are attributed to the slow continuous hydration of the sensing film and the transport of ionic species through the sensing film, which affects the overall Insulator capacitance. Thus, due to the transport of ionic species through the sensing film, we obtained a chemical modification of the gate insulator. The transport of ionic species is explained by a trap-limited mechanism.
transport and/or hopping mechanism known as dispersive diffusion.\textsuperscript{90} Dispersive transport is commonly observed in disordered materials, where the ion transport can arise due to trap-limited transport due to the presence of trap sites underneath the surface of the sensing film and/or hopping motion through localized states, which is reviewed in detail by Jamasb et al.\textsuperscript{46} This leads to the formation of a modified surface layer whose dielectric constant differs from the bulk insulator, where the diffusing species undergo dispersive transport across the modified surface layer formed previously in order to reach the insulator interface where further reaction occurs. Given the transport of ionic species is limited by the dispersive transport mechanism, the kinetics of growth of the modified surface layer follow a stretched-exponential time dependence in the presence of sites/traps.\textsuperscript{56} The thickness of the modified surface layer varies with time, which modifies the overall insulator capacitance. Thus, we obtained a temporal variation in the overall insulator capacitance, where a gradual reduction in the insulator capacitance causes monotonic variation in the ISFET threshold voltage, which gives rise to temporal drift in the drain current. Bousse et al. have reported that drift characteristics in Al₂O₃-gate ISFET pH sensors arise from bulk of the Al₂O₃ thin film and that it is not a surface phenomenon.\textsuperscript{45} Jamasb et al. have reported that the thickness of the chemically modified surface layer near the insulator–electrolyte interface is a few angstroms thick.\textsuperscript{91} The measured drift rates for different pH buffer solutions of 4, 7, and 10 are 0.136 μA/min, 0.124 μA/min, and 0.108 μA/min, respectively, as shown in Fig. 13. The obtained drift rates are better than TiN-gate ISFETs,\textsuperscript{88} which shows high hydration resistance of Al₂O₃ sensing film, which is in agreement with Jang and Cho’s study.\textsuperscript{92}

Hysteresis occurs due to the interaction of the surface of the sensing film with the ions present in the solution.\textsuperscript{45} Bousse et al. have reported that the imperfections in the Al₂O₃ sensing film cause some degree of porosity in the film, which creates interior sites. These buried sites interact slowly with the ions in the solution, as ionic diffusion needs time to reach the buried sites underneath the insulator surface,\textsuperscript{93} causing irreversible small changes in the drain current. The imperfections in the Al₂O₃ thin film may arise due to dissolution of the sensing film in pH buffer solutions.\textsuperscript{94} The hysteresis study is carried out for the pH cycle: pH 7 → pH 4 → pH 7 → pH 10 → pH 7 for a duration of 50 min. In order to carry out the measurements for hysteresis, the V$_{\text{Ref}}$ and V$_{\text{DS}}$ were fixed as 1 V each, and the current was recorded with a Keysight B2902A SMU. The hysteresis was measured as the change in the current from the initial to the final time duration in the pH cycle: pH 7 → pH 4 → pH 7 → pH 10 → pH 7. The obtained rate of hysteresis was nearly 5.806 μA, as shown in Fig. 14. We also calculated the acidic and basic rate of hysteresis. The acidic and the basic hysteresis values are measured separately for the pH cycle pH 7 → pH 4 → pH 7 and pH 7 → pH 10 → pH 7. The measured hysteresis values for acidic and basic solutions are nearly 4.401 μA and 1.405 μA, respectively. Therefore, the acidic and basic rate of hysteresis are obtained as 0.244 μA/min and 0.078 μA/min, respectively. The asymmetrical hysteresis of the pH sensor is due to the different diffusion rates of H$^+$ and OH$^-$ ions in the buried sites of the sensing film.\textsuperscript{95}

**Conclusions**

In this work, we have reported the fabrication, characterization, and modeling of an Al₂O₃-gate ISFET pH sensor. The ISFET sensor was fabricated through five levels of lithography using the self-aligned process. A pulsed-DC magnetron-assisted reactive sputtering process was used to deposit the sensing film, which was characterized by XRD, FESEM,
EDS, and XPS techniques to determine the film characteristics and surface morphology. Sensor characterization was performed in different pH buffer solutions, and the sensitivity was found to be nearly 42.1 mV/pH in the pH range 4–10. A behavioral macromodel of the sensor was built in SPICE to empirically extract various electrochemical parameters from the experimental results, and the experimental and simulated characteristics were found to be closely matching. Drift and hysteresis characteristics have also been studied, where the drift rates for different pH buffer solutions 4, 7, and 10 are 0.136 μA/min, 0.124 μA/min, and 0.108 μA/min, respectively, and a hysteresis of nearly 5.806 μA has been obtained. Thus, the developed sensing film is a promising material for chemical sensors and can be adopted for other popular applications of Al₂O₃ thin film, which include electrical insulation and abrasion protection.

In summary, the main contributions of the paper are as follows:

1. Optimization of the deposition process of Al₂O₃ sensing film using cost-effective pulsed-DC magnetron-assisted reactive sputtering technique and its integration with the fabrication of ISFET-based pH sensor.
2. Development of a facile fabrication process for an Al₂O₃-gate ISFET pH sensor using five steps of lithography, where the sensing film is patterned using the lift-off process.
3. Formulation of a SPICE macromodel for the fabricated Al₂O₃-gate ISFET pH sensor by incorporating the electrochemical parameters, which have been empirically obtained based on the experimental data.

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Conflict of interest The authors declare that they have no conflict of interest.

References

1. R. Edelstein, C. Tamanaha, P. Sheehan, M. Miller, D. Baselt, L. Whitman, and R. Colton, Biosens. Bioelectron. 14, 805 (2000).
2. J.-Y. Pyo, and W.-J. Cho, Semicond. Sci. Technol. 32, 035015 (2017).
3. K.S. Kim, H.-S. Lee, J.-A. Yang, M.-H. Jo, and S.K. Hahn, Nanotechnology 20, 235501 (2009).
4. H.-M. So, K. Won, Y.H. Kim, B.-K. Kim, B.H. Ryu, P.S. Na, H. Kim, and J.-O. Lee, J. Am. Chem. Soc. 127, 11906 (2005).
5. P. Bergveld, IEEE Trans. Biomed. Eng. 19, 342 (1972).
6. T.R. Martz, J.G. Connelly, and K.S. Johnson, Limnol. Oceanogr. Methods 8, 172 (2010).
7. C. Jimenez-Jorquera, J. Orozco, and A. Baldi, Sensors 10, 61 (2010).
8. N. Moser, T.S. Lande, C. Toumazou, and P. Georgiou, IEEE Sens. J. 16, 6496 (2016).
9. P. Bergveld, ISfet, theory and practice, in IEEE Sensors Conference, 1 (2003).
10. P. Bergveld, Sens. Actuators B Chem. 88, 1 (2003).
11. D. Sadighbayan, M. Hasanazadeh, and E. Ghafar-Zadeh, TrAC, Trends Anal. Chem. 133, 116067 (2020).
12. M. Kaisti, Biosens. Bioelectron. 98, 437 (2017).
13. M.W. Shinwari, M.J. Deen, and D. Landheer, Microelectron. Reliab. 47, 2025 (2007).
14. M.J. Schöning, and A. Poghossian, Analyst 127, 1137 (2002).
15. G. Geo, G. Lee, M.J. Kim, S.-H. Baek, M. Choi, K.B. Ku, C.-S. Lee, S. Jun, D. Park, H.G. Kim et al., ACS Nano 14, 5135 (2020).
16. A. Poghossian, M. Jablonski, D. Molinnus, C. Wege, and M.J. Schoening, Front. Plant Sci. 11, 1792 (2020).
17. S. Sinha, R. Mukhiya, R. Sharma, P. Khanna, and V. Khanna, J. Mater. Sci. Mater. Electron. 30, 7163 (2019).
18. R. Boidin, T. Halenković, V. Nazabal, L. Beneš, and P. Němec, Ceram. Int. 42, 1177 (2016).
19. G. Angarita, C. Palacio, M. Truijillo, and M. Arroyave, J. Phys. Conf. Ser. 850, 012022 (2017).
20. W. Ban, S. Kwon, J. Nam, J. Yang, S. Jang, and D. Jung, Thin Solid Films 641, 47 (2017).
21. A. Korcal, P. Płóciennik, A. Zawadzka, and B. Sahraoui, Optical properties of al 2 o 3 thin film deposited by sol-gel technique, in 2015 17th International Conference on Transparent Optical Networks (ICTON), IEEE, 2015, p. 1.
22. C. Barbos, D. Blanc-Pellissier, A. Fave, C. Botella, P. Regrene, G. Grenet, E. Blanquet, A. Crisci, and M. Lemit, Thin Solid Films 617, 108 (2016).
23. S. Jäger, B. Szyszka, J. Szczyrbowski, and G. Bräuer, Surf. Coat. Technol. 98, 1304 (1998).
24. J.D. Plummer, Silicon VLSI Technology: Fundamentals, Practice and Modeling (Pearson Education India, 2009).
25. V. Kirchhoff, T. Kopte, T. Winkler, M. Schulze, and P. Wiedemuth, Surf. Coat. Technol. 98, 828 (1998).
26. A. Belkind, Z. Zhao, D. Carter, L. Mahoney, G. McDonough, G. Roche, R. Scholl, and H. Walde, Pulsed-dc reactive sputtering of dielectrics: pulsing parameter effects, in Proceedings of the Annual Technical Conference-Society of Vacuum Coaters, 86 (2000).
27. P.J. Kelly, and R.D. Arnell, Vacuum 56, 159 (2000).
28. S. Rossnagel, Sputtering and sputter deposition, in Handbook of Thin Film Deposition Processes and Techniques, Elsevier, p. 319 (2001).
29. S. Schiller, K. Goedicke, V. Reshe, V. Kirchhoff, S. Schneider, and F. Milde, Surf. Coat. Technol. 61, 331 (1993).
30. P. Kelly, and J. Bradley, J. Optoelectron. Adv. Mater. 11, 1101 (2009).
31. J.A. Thornton, J. Vacuum Sci. Technol. A Vac. Surf. Films 4, 3059 (1986).
32. E. Tombach, A. Dobos, M. Szekeres, H. Narres, E. Klumpp, and I. Dekany, Colloid Polym. Sci. 278, 337 (2000).
33. E.O. Filatova, and A.S. Konashuk, J. Phys. Chem. C 119, 20755 (2015).
34. C. Cibert, H. Hidalgo, C. Champeaux, P. Tristant, C. Tixier, J. Desmaison, and A. Catherinot, Thin Solid Films 516, 1290 (2008).
35. P. Katiyar, C. Jin, and R. Narayan, Acta Mater. 53, 2617 (2005).
