LIONSIMBA: A Matlab Framework Based on a Finite Volume Model Suitable for Li-Ion Battery Design, Simulation, and Control

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The increasing demand for portable devices (e.g., smartphones) and hybrid electric vehicles (HEVs) calls for the design and management of storage devices of high power density and reduced size and weight. During the many decades of research, different chemistries of batteries have been developed, such as Nickel Cadmium (NiCd), Nickel Metal Hydride (NiMH), Lead Acid and Lithium ion (Li-ion) and Lithium ion Polymer (Li-Poly) (e.g., see Refs. 1–4). Among electrochemical accumulators, Li-ion batteries provide one of the best tradeoff in terms of power density, low weight, cell voltage, and low self-discharge.5 Mathematical models can support the design of new batteries as well as the development of new advanced battery management systems (ABMS).6–8 According to the literature, mathematical models for Li-ion battery dynamics fall within two main categories: Equivalent Circuit Models (ECMs) and Electrochemical Models (EMs). ECMs use only electrical components to model the dynamic behavior of the battery. ECMS include (i) the \( R_{\text{int}} \) model where only a resistance and a voltage source are used to model the battery, (ii) the \( RC \) model (introduced by the company SAF49) where capacitor dynamics have been added to the \( R_{\text{int}} \) model,10 and (iii) the Thvenin model, which is an extension of the \( RC \) model (e.g., see Refs. 11, 12 and references therein). In contrast, EMs explicitly represent the chemical processes that take place in the battery. While ECMs have the advantage of simplicity, EMs are more accurate due to their ability to describe detailed physical phenomena.13 The most widely used EM in the literature is the porous electrode theory-based pseudo-two-dimensional (P2D) model,14 which is described by a set of tightly coupled and highly nonlinear partial differential-algebraic equations (PDAEs). In order to exploit the model for simulation and design purposes, the set of PDAEs are reformulated as a set of ordinary differential-algebraic equations (DAEs). The model reformulation is very challenging to carry out in a way that is simultaneously computationally efficient and numerically stable for a wide range of battery parameters and operating conditions. To the authors’ best knowledge, no publication is available in the literature that provides a detailed step-by-step description of the numerical implementation of the P2D model or a freely available Matlab framework suitable for simulation, design, and development of ABMS for Li-ion batteries. In this article, starting from the P2D model, a computationally efficient and numerically stable finite volume DAE formulation is described in detail in order to facilitate implementation by the reader, while also addressing potential pitfalls and relative loopholes. Boundary conditions used to enforce physical meaningfulness of the system are thoroughly discussed and their numerical implementation is explained. Particular attention is directed to the handling of interface boundary conditions in the three primary sections of the battery - positive and negative electrodes and the separator. Due to possible discontinuities between adjacent sections, a mishandling of such conditions may lead to physically inconsistent solutions. Due to its intrinsic properties, the finite volume method has been chosen to easily deal with these particular interface conditions. Finally, based on the proposed finite volume discretization, we provide the Li-Ion SIMulation BAttery Toolbox (LIONSIMBA), a set of fully customizable Matlab functions suitable for simulating the dynamic behavior of Li-ion batteries. These functions are freely downloadable from the website http://sisdin.unipv.it/labsisdin/lionsimba.php. This article describes the features of our software. The user can implement his/her own custom-defined control algorithm to test different ABMS strategies, simulate cell behavior, optimize manufacturing parameters or test battery packs composed of series-connected cells. The package also allows the ready implementation of algorithms to estimate indexes such as the State of Charge (SOC) and the State of Health (SOH). The SOC is an important property of batteries that quantifies the amount of remaining charge (e.g., Ref. 15) and can be used to prevent damage, ensure safety, and minimize charging time.16 The SOH index measures the ability of the battery to store and deliver electrical energy; similar to the SOC, estimation-based approaches are used to predict the value of the SOH (e.g., see Refs. 17–19). The SOH tracks the long-term changes in a battery and its knowledge can help ABMS to anticipate problems through online fault diagnosis while providing charging profiles to slow down the battery aging. The package comes with the experimental parameters of the battery reported in Ref. 20. An initialization file allows changes in battery and simulator parameters. The simulator works under Matlab using IDA21 to solve the set of resulting DAEs with a good trade-off between accuracy and computational time.

The battery model and its numerical implementation is described first. Then the proposed framework is validated with respect to the results obtained using the COMSOL MultiPhysics commercial software22 and the Newman’s Fortran code named DUALFOIL.23 Finally, to demonstrate the effectiveness of the proposed software, thermal dynamics, model predictive control of state of charge, hybrid charge-discharge cycles, and a battery pack of series-connected cells are simulated. The toolbox is equipped with all the Matlab source files able to reproduce the simulations presented in this article.

Battery Model

The P2D model consists of coupled nonlinear PDAEs for the conservation of mass and charge in the three sections of

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battery – cathode, separator, and anode – denoted respectively by the indexes p, s, and n. The positive and negative current collectors are denoted by a and z. The index i ∈ S is used to refer to a particular section of the battery, where S = {a, p, s, n, z}. All model equations are reported in Tables I and II. Variables c_r(x, t), c_s(x, t), and c_a(x, t) ∈ ℝ⁺ denote the electrolyte concentration, the average concentration in the solid particles, and the surface concentration in the solid particles of Li-ions respectively, where time t ∈ ℝ⁺ and x ∈ ℝ is the spatial direction along which the ions are transported. Diffusion inside solid spherical particles with radius R_p is described by Fick’s law,
\[
\frac{\partial c_{r}(x, t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left[ r^2 D_{p} \frac{\partial c_{r}(x, t)}{\partial r} \right],
\]
with boundary conditions
\[
\frac{\partial c_{r}(x, t)}{\partial r} \bigg|_{r=0} = 0, \quad \frac{\partial c_{r}(x, t)}{\partial r} \bigg|_{r=R_p} = -j(x, t) D_{p}^{-1},
\]
where r is the radial direction along which the ions intercalate within the active particles. This model introduces a pseudo-second dimension (r). To reduce complexity and computational burden, Refs. 24 and 25 proposed different efficient reformulations for the solid-phase diffusion equation. As discussed in Ref. 26, according to the particular application, different model reformulations can be employed while maintaining good accuracy. For low to medium C rates, the diffusion length method²⁷ or the two-term polynomial approximation method are accurate. At high C rates, higher-order polynomial approximations or the Pseudo Steady State (PSS)²⁸ approximation can be employed.

For more details, refer to Ref. 26 and the references therein.

In the two-term polynomial approximation, concentration profiles inside the particle are assumed to be quadratic in r and 1 is approximated by means of average and surface concentration of the solid particles,
\[
\frac{\partial c_{s}^{avg}(x, t)}{\partial t} = -3 \frac{j(x, t)}{R_p},
\]
\[
c_{r}^{p}(x, t) - c_{r}^{avg}(x, t) = \frac{R_p j(x, t)}{D_{p}^{-1}}.
\]
This reformulation leads to a one-dimensional problem in x by removing the pseudo-second dimension r. Despite the reduced computational burden, such approximation could lead to a decrease of the prediction accuracy for high rates, short time responses or pulse currents.²⁶ For these applications, higher-order polynomials or Fick’s law are recommended, as discussed in Solid-phase diffusion models section.

The electrolyte and solid potential are represented by \( \Phi_s(x, t) \) and \( \Phi_r(x, t) \) ∈ ℝ, while \( T(x, t) \) and \( j(x, t) \) represent the temperature and the ionic flux. Note that the ionic flux is present only in the positive and negative electrodes, and not in the separator. The open circuit voltage (OCV) is denoted by \( U \) while the entropic variation of the OCV is denoted by \( \frac{\partial U}{\partial T} \).

The cathode, anode, and separator are composed of different materials: for a given section i, different electrolyte diffusion coefficients \( D_i \), solid-phase diffusion coefficients \( D_i^s \), electrolyte conductivities \( \kappa_i \), porosities \( \epsilon_i \), thermal capacities \( C_{p,i} \), thermal conductivities \( \lambda_i \), densities \( \rho_i \), solid-phase conductivities \( \sigma_i \), particle surface area to volumes \( a_i \), maximum solid phase concentrations \( c_i^{max} \), overpotentials \( \eta_i \), and particle radiiuses \( R_p \) can be defined. The terms \( R \) and \( F \) are the universal gas constant and the Faraday constant, respectively, with \( \tau_i \) representing the transference number. The applied current density is \( I_{app}(t) \), and \( T_{env} \) denotes the environment temperature. In order to take into account the properties of different materials

### Table I. Li-ion P2D model governing equations.

| Current Collectors, \( i \in \{a, z\} \) | Boundary Conditions |
|------------------------------------------|---------------------|
| \( \rho_i C_p \frac{\partial T_i(x, t)}{\partial t} = \frac{\partial}{\partial t} \left[ \lambda_i \frac{\partial T_i(x, t)}{\partial r} \right] + \frac{\partial}{\partial r} \left[ \frac{1}{\sigma_{i \tau}} \frac{\partial T_i(x, t)}{\partial r} \right] \) | \( -\lambda_c \frac{\partial T_c(x, t)}{\partial x} \bigg|_{x=0} = h(T_{env} - T(x, t)) \) |
| Positive and Negative Electrodes, \( i \in \{p, n\} \) | \( -\lambda_c \frac{\partial T_c(x, t)}{\partial x} \bigg|_{x=L} = h(T(x, t) - T_{env}) \) |

\( c_i^p(x, t) - c_i^{avg}(x, t) = -R_p j(x, t) \) and \( a_i \) define the active particles. This model introduces a pseudo-second dimension (r). To reduce complexity and computational burden, Refs. 24 and 25 proposed different efficient reformulations for the solid-phase diffusion equation. As discussed in Ref. 26, according to the particular application, different model reformulations can be employed while maintaining good accuracy. For low to medium C rates, the diffusion length method²⁷ or the two-term polynomial approximation method are accurate. At high C rates, higher-order polynomial approximations or the Pseudo Steady State (PSS)²⁸ approximation can be employed.

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\frac{\partial c_{i}^{avg}(x, t)}{\partial t} = -3 \frac{j(x, t)}{R_p},
\]
\[
c_{i}^{p}(x, t) - c_{i}^{avg}(x, t) = \frac{R_p j(x, t)}{D_{p}^{-1}}.
\]
This reformulation leads to a one-dimensional problem in x by removing the pseudo-second dimension r. Despite the reduced computational burden, such approximation could lead to a decrease of the prediction accuracy for high rates, short time responses or pulse currents.²⁶ For these applications, higher-order polynomials or Fick’s law are recommended, as discussed in Solid-phase diffusion models section.

The electrolyte and solid potential are represented by \( \Phi_s(x, t) \) and \( \Phi_r(x, t) \) ∈ ℝ, while \( T(x, t) \) and \( j(x, t) \) represent the temperature and the ionic flux. Note that the ionic flux is present only in the positive and negative electrodes, and not in the separator. The open circuit voltage (OCV) is denoted by \( U \) while the entropic variation of the OCV is denoted by \( \frac{\partial U}{\partial T} \).

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Table II. Additional equations.

Open Circuit Potential (Thermal dependence)

\[
U_p = U_{p,ref} + (T(x, t) - T_{ref}) \frac{\partial U_p}{\partial T}_{T_{ref}}
\]

\[
U_n = U_{n,ref} + (T(x, t) - T_{ref}) \frac{\partial U_n}{\partial T}_{T_{ref}}
\]

Entropy change

\[
\left. \frac{\partial U_p}{\partial T} \right|_{T_{ref}} = -0.001 \left( 0.199529039 - 0.928373828 + 1.364550689 x 0.00039 - 0.611544903999999 \right) \]

\[
\left. \frac{\partial U_n}{\partial T} \right|_{T_{ref}} = 0.001 \left( 0.005269056 + 3.299265709 - 91.79325798 \theta_0^{3} + 1004.911008 \theta_0^{5} - 5812.278127 \theta_0^{7} + 19329.7549 \theta_0^{8} - 37147.89476 \theta_0^{9} + 38379.18127 \theta_0^{10} - 16515.05308 \theta_0^{11} \right)
\]

Open Circuit Potential (Reference value)

\[
U_{p,ref} = -4.656 + 88.660 \theta_{j}^{0} - 401.119 \theta_{j}^{2} + 342.509 \theta_{j}^{4} - 462.471 \theta_{j}^{6} + 433.434 \theta_{j}^{8} - 95.96 \theta_{j}^{10}
\]

\[
U_{n,ref} = 0.7222 + 0.1387 \theta_{a} + 0.029 \theta_{a}^{3} - \frac{0.0172}{\theta_{a}} + \frac{0.0019}{\theta_{a}^{3}} + 0.2808 e^{0.9 - 15 \theta_{a}} - 0.7984 e^{0.446 \theta_{a}} - 0.4108
\]

\[
\theta_{p} = \frac{c_{p}^{e}(x, t)}{c_{p}^{e, ref}}
\]

\[
\theta_{a} = \frac{c_{a}^{s}(x, t)}{c_{a}^{s, ref}}
\]

Heat source terms (Anode and Cathode)

\[
Q_{ohm} = \sigma_{eff,i} \left( \frac{\partial \Phi_{i}(x, t)}{\partial x} \right) + \kappa_{eff,i} \left( \frac{1}{\theta_{i}(x, t)} \right) \frac{\partial \Phi_{i}(x, t)}{\partial x}, \quad i \in \{p, n\}
\]

\[
Q_{an} = F a_{i} j(x, t) \eta_{i}(x, t), \quad i \in \{p, n\}
\]

\[
Q_{rev} = F a_{i} j(x, t) T(x, t) \frac{\partial U_i}{\partial T}_{T_{ref}} , \quad i \in \{p, n\}
\]

Heat Source terms (Separator)

\[
Q_{ohm} = \omega_{eff,i} \left( \frac{\partial \Phi_{i}(x, t)}{\partial x} \right) + 2 \pi \frac{\sigma_{eff,i} \partial \Phi_{i}(x, t)}{\partial x} \left( 1 - T_{ref} \right) \frac{\partial \ln c_{i}(x, t)}{\partial T_{x}} \frac{\partial \Phi_{i}(x, t)}{\partial x}, \quad i = s
\]

Various Coefficients

\[
D_{eff,i} = k_{i} \times 10^{-4} \times 10^{-4.4} \times \frac{1}{T_{x}} \times 10^{-10} \times c_{i}(x, t)
\]

\[
\kappa_{eff,i} = k_{i} \times 10^{-4} \times c_{i}(x, t)
\]

\[
\eta_{i} = c_{i} \left( 1 - \theta_{i} \right) \frac{\partial c_{i}(x, t)}{\partial x}
\]

\[
\gamma = \frac{2 \theta_{i} - \theta_{i}}{\theta_{i} - \theta_{i}}
\]

used in the battery, effective diffusion and conductivity coefficients are evaluated according to the Bruggeman’s theory, with “eff” suffixes representing effective values of such coefficients. The thickness of the overall battery is L, where L = \sum L_i and L_i represents the length of each battery section. Due to physical constraints, it is necessary to impose (i) zero-flux boundary conditions for the \( c_i \) diffusion equation at the two ends of the battery, (ii) Newton’s cooling law for the dissipation of heat in the system, and (iii) null-flux conditions for \( \Phi_i \) at the interface between electrodes and the separator as well as the enforcement of Ohm’s law at the end of the electrodes. Given that only potential differences are measurable, without loss of generality, \( \Phi_i \) can be set to zero at the end of the anode. Similarly, on the cathode side, zero-flux conditions are imposed. Within the battery, interface conditions are imposed across the different materials. In order to get a more detailed description of the conductivity and diffusion phenomena inside the electrolyte, all the related coefficients are determined as a function of \( \theta \) and \( T \), as discussed in Ref. 29.

Excessive heat generation may lead to performance degradation and, in extreme cases, thermal runaway of the cell.30,31 In order to address these possible safety issues, thermal dynamics are included with the set of conservation equations describing the thermal. The thermal equations include different source terms, which are the ohmic, reversible, and reaction generation rates \( Q_{ohm} \), \( Q_{ev} \), and \( Q_{an} \), respectively.32 The ohmic generation rate takes into account heat
generated as a consequence of the motion of Li-ions in the solid/liquid phase. The reaction generation rate accounts for heat generated due to ionic flux and over-potentials, and the reversible generation rate takes into account the heat rise due to the entropy change in the electrodes’ structure. The next section uses the notation $\dot{\bar{x}}_a = I_a, \dot{\bar{x}}_p = I_a + I_p, \dot{\bar{x}}_s = I_a + I_p + I_s$, and $\dot{\bar{x}}_h = I_a + I_p + I_s + I_h$. For a clearer comprehension, bold is used in tables for coefficients whose dependence on other variables is made explicit in other equations. The nomenclature of the variables is reported in List of symbols section. The model equations are from Ref. 14, where for convenience the electrolyte other variables is made explicit in other equations. The nomenclature sent the infinitesimal volume of prescribed spatial domain.

**Numerical Implementation**

Most numerical methods for model-based estimation and control algorithms require the model to be formulated in terms of AEs or DAEs rather than PDAEs. Different numerical methods can be used to achieve this objective. The reformulation process from PDAEs to AEs or DAEs is carried out by discretizing the domains of the independent variables (e.g., the time domain $t$ and the $n$-dimensional spatial domain $x \in \mathbb{R}^n$). The discretization can involve both time and space, to produce AEs, or only space, to produce DAEs. An example of discretization in time and space is given by the FTCS (Forward-Time Central-Space) approach. Other techniques, like the method of lines (MOL), discretize only the space domain and leave the time as a continuous variable. When this latter approach is used, finite volume, finite difference, or finite element methods can be employed to obtain the set of DAEs. Alternative approaches can be used. For example, orthogonal collocation can be used with an efficient coordinate transformation to solve the set of resulting DAEs. In this paper, in order to exploit the properties of variable-step solvers, MOL is used to reformulate the original set of PDAEs. In particular, the finite volume method (FVM) is employed. Due to its ability to conserve properties with high accuracy (within numerical roundoff errors), the FVM has been used in literature to discretize models in a wide range of applications, such as heat transfer problems, flow and transport in porous media, or more general applications for hyperbolic problems as discussed in Ref. 39. In particular, the FVM together with the harmonic mean (HM) have been used to deal with possible discontinuities across different sections of the cell. To the authors’ knowledge, no published work addresses in detail the numerical issues related to the implementation of the Li-ion cell model and, in particular, the handling of boundary conditions that ensure physical meaningfulness of the obtained solutions. For this reason, all the numerical details are addressed below.

**Finite volume formulation.**—Consider a general diffusion-convection equation defined on a domain $\Omega \subset \mathbb{R}^n$ of the form

$$\frac{\partial \phi}{\partial t} + \nabla (\eta \phi) = \nabla (\Gamma \nabla \phi) + s$$

where $\phi$ is the unknown variable, $\eta$ is the velocity, $\Gamma$ is a diffusion coefficient and $s$ a source term. Both the unknown $\phi$ and the source term $s$ depend on time $t$ and space $x \in \mathbb{R}^n$. For convenience define $f(\phi) := \eta \phi - \Gamma \nabla \phi$. Integrating 2 over a spatial domain $\Omega \subset \mathbb{R}^n$ and applying the divergence theorem produces the integral form of the conservation law:

$$\int_{\Omega} \frac{\partial \phi}{\partial t} dV + \int_{\partial \Omega} (f(\phi) \cdot n) dS = \int_{\Omega} s dV$$

where $d\Omega$ is the boundary of the domain $\Omega$, $n$ is the outward pointing unit normal on the boundary of the domain, and $dV$ and $dS$ represent the infinitesimal volume of $\Omega$ and the infinitesimal surface of the boundary $\partial \Omega$ respectively. Alternatively, this integral equation could be written directly as an exact conservation equation over any prescribed spatial domain.

![Figure 1](image1.png)  
**Figure 1.** Example of a 2D FVM mesh where the set of neighbor cells $C(k)$ is represented by the green cells.

According to the FVM, the spatial domain $\Omega$ is divided into a set of disjoint control volumes (CVs) $\Omega_k$ centered in $x_k \in \mathbb{R}^n$, such that $\Omega = \bigcup \Omega_k$ and $\Omega_i \cap \Omega_j = \emptyset, \forall i \neq j$. The thermal model is taken from Ref. 32 while all of the parameters describing the particular chemistry are taken from Ref. 20.

**Discretization of the governing equations.**—The discretization method introduced in Finite volume formulation section is exploited to reformulate the set of governing equations summarized in Table 1. Given that all the unknowns of the Li-ion cell model are functions of the variables $t \in \mathbb{R}^+$ and $x \in \mathbb{R}$, the development of a 1D FVM model is addressed. In order to correctly carry out the discretization process, a mesh structure is defined by subdividing the spatial domain into $N_a + N_p + N_s + N_h$ non-overlapping volumes with geometrically centered nodes (as depicted in Fig. 2). Every CV is associated with a

![Figure 2](image2.png)  
**Figure 2.** One-dimensional finite volume mesh.
Table III. FVM P2D equations.

| Current Collectors, \(i \in \{a, z\} \) | Boundary Conditions |
|-----------------------------------------|---------------------|
| (T) \( \rho_i C_{p,i} \frac{\partial \theta_i}{\partial t} = \frac{1}{\Delta V} \left[ \lambda_i \frac{\partial \theta_i}{\partial x} \right]_{x_{k-\frac{1}{2}}}^{x_{k+\frac{1}{2}}} + \frac{i_{app,i}}{\sigma_{f,i} \rho_i} \) | \( \frac{\partial T(t)}{\partial x} \bigg|_{x_0} = h(T_{ref} - T(t)) \) |
| Positive and Negative Electrodes, \(i \in \{p, n\} \) | \( \frac{\partial T(t)}{\partial x} \bigg|_{x_n} = h(T_{end}(t) - T_{ref}) \) |

(M1) \( \epsilon \frac{\partial \Phi_{i}(x,t)}{\partial t} = \frac{1}{\Delta V} \left[ D_{eff,i} \frac{\partial \Phi_{i}(x,t)}{\partial x} \right]_{x_{k-\frac{1}{2}}}^{x_{k+\frac{1}{2}}} + a_i (1 - e^{-\frac{\epsilon}{\Delta \epsilon}}) j_i(t) \)

(M2) \( \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_0} = -\frac{3 \Phi_{i}(x,t)}{\rho_i \sigma_{f,i}} \)

(M3) \( \frac{\partial \Phi_{i}(x,t)}{\partial x} - \frac{3 \Phi_{i}(x,t)}{\rho_i \sigma_{f,i}} = \frac{E_{app,i}}{\rho_i \sigma_{f,i}} \)

(C1) \( \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx a_i F \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} = \Delta \epsilon \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \)

(C2) \( \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} - \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} = \Delta \epsilon \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} = 0 \)

(T) \( \rho_i C_{p,i} \frac{\partial T(t)}{\partial t} = \frac{1}{\Delta V} \left[ \lambda_i \frac{\partial T(t)}{\partial x} \right]_{x_{k-\frac{1}{2}}}^{x_{k+\frac{1}{2}}} + Q_{source,k} \)

\( \frac{\partial T(t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} + Q_{source,k} \)

Separator, \(i = z\)

(M1) \( \epsilon \frac{\partial \Phi_{z}(x,t)}{\partial t} = \frac{1}{\Delta V} \left[ D_{eff,z} \frac{\partial \Phi_{z}(x,t)}{\partial x} \right]_{x_{k-\frac{1}{2}}}^{x_{k+\frac{1}{2}}} \)

(C2) \( \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} - \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} = \Delta \epsilon \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} = 0 \)

\( \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} + Q_{shunt,k} \)

whereas the derivatives present in the ohmic source are numerically approximated as

\( \frac{\partial \Phi_{i}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\Phi_{i,k+1}(t) - \Phi_{i,k-1}(t)}{2 \Delta x_i} \)

\( \frac{\partial \Phi_{z}(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\Phi_{z,k+1}(t) - \Phi_{z,k-1}(t)}{2 \Delta x} \)

\( \frac{\partial \ln c_i(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{c_{i,k+1}(t) - c_{i,k-1}(t)}{2 \Delta x_i c_{i,k}} \)

using a central differencing scheme. Finally the term \( \Phi_{source,k} := Q_{shunt,k} + Q_{rev,k} + Q_{run,k} \).

Equation (C2) in Table III requires the evaluation of \( T(x,t), c_i(x,t) \), and \( k_{eq} \) at the edges of the CVs. For example, consider Fig. 3, where the value of the unknown \( T \) has been evaluated at the interface between two CVs. In order to recover such value, linear interpolation techniques are used. The same approach is also applied for \( c_i \) and \( k_{eq} \).

As discussed in Finite volume formulation section, a suitable numerical approximation for \( F(\Phi) \) is needed. Given that no convective terms are present in the set of governing equations, numerical approximation is only required for the diffusive terms (e.g., \( -\nabla \Phi \)). In this work, all the diffusive terms are numerically approximated with a first-order scheme:

\( \frac{\partial \Phi(x,t)}{\partial x} \bigg|_{x_{k-\frac{1}{2}}} \approx \frac{\Phi(x,t) - \Phi_{k-1}(t)}{\Delta x} \)

\( \frac{\partial \Phi(x,t)}{\partial x} \bigg|_{x_{k+\frac{1}{2}}} \approx \frac{\Phi(x,t) - \Phi_{k}(t)}{\Delta x} \)

All the values coming from the additional equations in Table II are obtained as a function of the average values of the unknowns. Equation (T) is used to obtain the values of \( T \), while equations (M1), (M2), and (M3) are used to obtain the values of \( c_i \), \( c_i^n \), and \( c_i^\prime \) respectively.

Figure 3. Interpolation technique to recover edge values of the unknowns.
Implementation of boundary and interface conditions.—Boundary conditions must be enforced to have a physically meaningful solution. As shown in Table I, null-flux boundary conditions on the electrolyte diffusion equation $c_i$ can be straightforwardly enforced by imposing $z_i = 0$ at $x = \hat{x}_0$ and $x = \hat{x}_s$. The same procedure can be used to enforce $\frac{\partial z_i}{\partial x} = 0$ at $x = \hat{x}_0$, while $\Phi_s = 0$ at $x = \hat{x}_s$, is enforced by setting to zero the value of $\Phi_s$ at the last CV of the anode. Solid-phase potential boundaries are enforced by substituting $\frac{\partial \Phi_s}{\partial x}$ at $x = \hat{x}_0$ and $x = \hat{x}_s$ the value of $-\frac{\partial \Phi_s}{\partial x}$, respectively. Similarly, at $x = \hat{x}_p$ and $x = \hat{x}_s$, $\frac{\partial \Phi_s}{\partial x}$ is replaced by the value 0. To enforce heat exchange with the surrounding environment, the terms $\frac{\partial T}{\partial x}$ evaluated at $x = 0$ and $x = L$ are substituted with the terms $h(T_{int} - \bar{T})$ and $-h(T_{int} - T_{el})$ respectively. The suffixes $i$ and $el$ refer to the first and last CV of the entire mesh. All these conditions have been formulated also for the FVM discretization as shown in Table III.

Due to changes in material properties along the length of the battery, interface conditions are required to enforce continuity of the solution. For this reason, the values of different coefficients (e.g., $D_{eff,i}$, $\kappa_{eff,i}$, $\lambda_i$) need to be evaluated at the interface between two different materials. The easiest way would be to use an arithmetic mean; however, in some cases, this approach cannot accurately handle the abrupt changes of coefficients that may occur. Instead, the HM is employed to evaluate the value at the edges of the CVs. The HM of two generic coefficients ($k_1$ and $k_2$) can be expressed as

$$\frac{k_1 k_2}{\beta k_2 + (1 - \beta) k_1}$$

where $\beta$ represents a weight to account for the difference between the different CV widths. A common value for $\beta$ is $\beta = \frac{\Delta x_1}{\Delta x_1 + \Delta x_s}$, where $\Delta x_1$ and $\Delta x_s$ represent the CV widths. This formulation produces results that are more robust in presence of the abrupt changes of the coefficients, without requiring an excessively fine grid in the vicinity of the interface.

Consider Fig. 4 where the interface across the last volume of the cathode and the first volume of the separator is depicted. Remember that, as discussed in Discretization of the governing equations section, the mesh structure has been chosen in order to align the CV edges with the interfaces or physical boundaries of the battery. The value of $D_{eff,i+\frac{1}{2}}$ can be obtained using the HM as

$$D_{eff,i+\frac{1}{2}} = \frac{D_{eff,i} D_{eff,i+1}}{\beta D_{eff,i+1} + (1 - \beta) D_{eff,i}}$$

where $\beta = \frac{\Delta x_1}{\Delta x_1 + \Delta x_s}$. The electrolyte diffusion in the last volume of the cathode is

$$\epsilon_p \frac{\partial \bar{c}_i(t)}{\partial t} = D_{eff,i+\frac{1}{2}} \frac{(\bar{c}_i(t) - \bar{c}_i(t))}{\Delta x_{i+\frac{1}{2}}} - D_{eff,i-\frac{1}{2}} \frac{(\bar{c}_i(t) - \bar{c}_i(t))}{\Delta x_{i-\frac{1}{2}}} + a_p (1 - t_i) \bar{h}(t)$$

whereas

$$\epsilon_s \frac{\partial \bar{c}_s(t)}{\partial t} = D_{eff,i+\frac{1}{2}} \frac{(\bar{c}_s(t) - \bar{c}_s(t))}{\Delta x_{i+\frac{1}{2}}} - D_{eff,i-\frac{1}{2}} \frac{(\bar{c}_s(t) - \bar{c}_s(t))}{\Delta x_{i-\frac{1}{2}}}$$

in the first volume of the separator, with $\Delta \bar{x} = \frac{\Delta x_1 + \Delta x_s}{2}$. The same approach is used to enforce interface conditions where needed.

When dealing with battery packs, in particular with series-connected cells, all the aforementioned numerical schemes have to be replicated for each cell. Moreover, when temperature dynamics are considered, the numerical scheme has to be adapted in order to account for continuity fluxes across the cells. Indeed, if two cells are connected in series,

$$-\lambda_{c,1} \frac{\partial T_i(x,t)}{\partial x} \bigg|_{x=x^e} = -\lambda_{c,2} \frac{\partial T_i(x,t)}{\partial x} \bigg|_{x=x^e}$$

must hold at the interface of the current collectors across the two cells (e.g., at $x = x^e$), where $T_i(x,t)$ is the temperature of the current collector of the $i$th cell. Finally, Newton’s law of cooling has to be enforced respectively at the positive current collector of the first cell and at negative current collector of the second cell.

**Li-ion Simulation Battery Toolbox (LIONSIMBA)**

Different implementations of Li-ion cell simulation have been reported in the literature written in such languages as Maple and Fortran (DUALFOIL), and in numerical analysis commercial software such as COMSOL Multiphysics and Modelica which provide a variety of models to simulate the behavior of a Li-ion cell. Matlab, however, is the software language most commonly used by researchers for the development and evaluation of different identification, estimation, and control algorithms, as Matlab has by far the largest number of toolboxes that implement the widest variety of such algorithms. Combined with its Instrument Control Toolbox that has a very extensive suite of protocols for directly communicating and controlling laboratory equipment, Matlab has the maximum flexibility for evaluation of control algorithms through simulations and experiments.

Based on the aforementioned Li-ion cell model, this work provides a freely available Matlab based software for the simulation of Li-ion cells, Li-ion SIMulation Battery Toolbox (LIONSIMBA), to serve as a reference simulation environment for the facile development and evaluation of different ABMs. Due to its native integration with the Matlab environment, the software facilitates the development of new algorithms, such as for the identification of Li-ion cell parameters, State of Charge, and optimal charging. LIONSIMBA is freely downloadable at: http://sisdin.unipv.it/labsisdin/lionsimba.php.

The package comes with different Matlab editable.m scripts:

- **electrolyteDiffusionCoefficients.m**: computes the electrolyte diffusion coefficients.
- **electrolyteConductivity.m**: computes the electrolyte conductivity coefficients.
- **openCircuitPotential.m**: used to compute the Open Circuit Potential (OCP).
- **reactionRates.m**: computes the reaction rate coefficients for the ionic flux.
- **solidPhaseDiffusionCoefficients.m**: computes the solid phase diffusion coefficients.

This set of scripts refer to version 1.02 of the software; modifications or other scripts can be added in future releases of the software.
The numerical implementation of the LIONSIMBA has been carried out according to the rules outlined in Numerical Implementation section and the cell considered is a LiCoO₂ and LiC₆ system. All the parameter values have been taken from the real battery data in Ref. 20, and are summarized in Table IV.

**LIONSIMBA validation.**—The P2D model has been experimentally validated numerous times since, this section validates the numerical implementation of LIONSIMBA by comparing the results coming from the proposed framework with COMSOL MultiPhysics and DUALFOIL. While COMSOL has been supplied with the same model used in our framework, where a heat diffusion PDE is used to describe the thermal dynamics, DUALFOIL neglects the spatial distribution of the temperature and averages the heat generation rates over the cell. For this reason, the comparison among the three different codes is carried out considering isothermal conditions. The thermal model is included in the comparison with COMSOL. For isothermal and thermal enabled scenarios, a 1C discharge cycle is performed, while the same set of parameters are maintained across the different codes.

The cell potentials \( V(t) \), electrolyte concentrations \( c_s(x,t) \), potential \( \phi_s(x,t) \), and surface solid-phase concentrations \( c_s'(x,t) \) for the isothermal battery are nearly identical for LIONSIMBA, COMSOL, and DUALFOIL (see Fig. 5). For the thermal enabled scenario, the cell potentials, temperature, and other internal states for LIONSIMBA are nearly identical to COMSOL (see Fig. 6).

**Solid-phase diffusion models.**—As introduced in Battery model section, according to the P2D model developed in Ref. 14, diffusion inside the solid particles is described using Fick’s law, where the presence of a second-pseudo dimension \( r \) can significantly increase the computational burden. According to the particular application under study, different approximations of 1 can be employed without significant loss of accuracy. The choice of the solid-phase diffusion model should be cautious, as approximate models can have poor accuracy in scenarios comprising high rate of charge/discharge, short time simulations, or pulse currents. For this reason, LIONSIMBA allows the user to choose among three different models for solid-phase diffusion:

- Fick’s law (including the pseudo-second dimension \( r \)):
  \[
  \frac{\partial c_s(x,t)}{\partial t} = \frac{1}{r^2} \frac{\partial}{\partial r} \left( r^2 D_{eff} \frac{\partial c_s(r,t)}{\partial r} \right)
  \]
  with boundary conditions
  \[
  \frac{\partial c_s(r,t)}{\partial r} \bigg|_{r=0} = 0, \quad \frac{\partial c_s(r,t)}{\partial r} \bigg|_{r=R_p} = -\frac{j(x,t)}{D_{eff}}
  \]
  - two-parameter polynomial approximation: \[ c_s'(x,t) - c_s^{pol}(x,t) = -\frac{R_p}{D_{eff}} \frac{j(x,t)}{5} \]
  - higher-order polynomial approximation: \[ c_s'(x,t) - c_s^{pol}(x,t) = -\frac{j(x,t)}{R_p} + \frac{45}{2} \frac{j(x,t)}{R_p} q(x,t) + 8 \frac{j(x,t)}{35 D_{eff}} + 8 R_p q(x,t) \]

The prediction accuracy of each approximate model is assessed by comparison of the cell potential vs. time profiles for different C rates for Fick’s law with the two approximate models (see Fig. 7). The two-parameter approximation accurately describes the cell potential for low to medium C rates (1C to 2C, Figs. 7a and 7b) and the higher order polynomial approximation is accurate up to the 5C rate.
Figure 5. Validation of the LIONSIMBA numerical implementation in isothermal conditions, with the legend given in part a. (a) Cell potential. (b) Electolyte Li-ion concentration. (c) Solid phase Li-ion concentration. (d) Electrolyte potential.

Table IV. List of parameters used in simulation.20

| Parameter | Description | Aluminium CC | Cathode | Separator | Anode | Carbon CC |
|-----------|-------------|--------------|---------|-----------|-------|-----------|
| $c_{\text{li,ini}}$ | Initial concentration in the electrolyte | – | 1000 | – | 1000 | – |
| $c_{\text{li,max}}$ | Maximum solid-phase concentration | – | 51554 | – | 30555 | – |
| $D_r$ | Electrolyte diffusivity | – | $7.5 \times 10^{-10}$ | $7.5 \times 10^{-10}$ | $7.5 \times 10^{-10}$ | – |
| $D_r'$ | Solid-phase diffusivity | – | $10^{-14}$ | – | $3.9 \times 10^{-14}$ | – |
| $k_i$ | Reaction rate constant | – | $2.334 \times 10^{-11}$ | – | $5.031 \times 10^{-11}$ | – |
| $l_p$ | Thickness | 10$^{-5}$ | $8 \times 10^{-5}$ | $2.5 \times 10^{-5}$ | $8.8 \times 10^{-5}$ | $10^{-5}$ |
| $R_{\text{p,i}}$ | Particle radius | – | $2 \times 10^{-6}$ | – | $2 \times 10^{-6}$ | – |
| $\rho_i$ | Density | – | 2500 | 1100 | 2500 | 8940 |
| $C_{\text{p,i}}$ | Specific heat | – | 897 | 700 | 700 | 385 |
| $\lambda_i$ | Thermal conductivity | – | 237 | 2.1 | 0.16 | 1.7 | 401 |
| $\sigma_i$ | Solid-phase conductivity | 3.55 $\times 10^7$ | 100 | 100 | 5.96 $\times 10^7$ | – |
| $\epsilon_i$ | Porosity | – | 0.385 | 0.724 | 0.485 | – |
| $\rho_i$ | Particle surface area to volume | – | 885,000 | – | 723,600 | – |
| $E_{\text{ki}}$ | Solid-phase diffusion activation energy | – | 5000 | – | 5000 | – |
| $E_{\text{ki}}'$ | Reaction constant activation energy | – | 5000 | – | 5000 | – |
| $\text{brugg}$ | Bruggeman’s coefficient | – | 4 | 4 | 4 | – |
| $F$ | Faraday’s constant | 96485 | – | – | – | – |
| $R$ | Universal Gas constant | 8.314472 | – | – | – | – |
| $\lambda_{\text{h}}$ | Transference number | 0.364 | – | – | – | – |
| $\epsilon_{f,i}$ | Filler fraction | – | 0.025 | – | 0.0326 | – |

The performance of each approximate model are quantified by root-mean-square error (RMSE) and normalized time index (NTI) in Table VI, where the RMSE is evaluated by comparing an approximate model solution with respect to the full model, while the normalized time index is the ratio between the computational time required by an approximate model and the time required by the full model with Fick’s law to simulate different scenarios. The two-term polynomial approximation has much less than 1% error for the 1C and 2C rates, with more than 1% error for higher rates. In all of the scenarios, this approximate model takes $\approx 80\%$ less time than the full model to simulate the cell.

The higher-order polynomial approximation has a factor of 4 to 5 lower RMSE for each scenario than for the two-term model, but an increase in computational time by a factor of two or more due to the inclusion of another set of PDEs. Although the RMSE of 1.9% at 10C could be small enough for some applications, the reduction in computational time is only about 37% compared to solving the full Fick’s law model.

Simulations

Simulation results were obtained using Matlab R2014b on a Windows 7@3.2GHz PC with 8 GB of RAM for the experimental battery parameters in Ref. 20 with a cutoff voltage of 2.5 V and environmental temperature of 298.15 K. For the proposed chemistry, the 1C value is $\approx 30$ A/m$^2$. The effectiveness and ease of use of the proposed framework are shown in a series of simulations.

In the first scenario (Fig. 8), 1C discharge simulations are compared for a very wide range of heat exchange coefficient $h$, with high $h$ being the most challenging for retaining numerical stability in dynamic simulations. As expected, decreasing the value of the parameter $h$ leads to a faster increase of the cell temperature. Moreover, due to the coupling of all of the governing equations, it is possible to note the influence of different temperatures on the cell voltage.

In the second scenario (Fig. 9), for a fixed value of $h = 1$ W/(m$^2$K), different discharge cycles are compared at 0.5C, 1C, and 2C. According to the different applied currents, the temperature rises in different ways; it is interesting to note the high slope of the temperature during a 2C discharge, mainly due to the electrolyte concentration $c_e$ being driven to zero in the positive electrode by the high discharge rate.
Figure 6. Validation of the LIONSIMBA numerical implementation with thermal dynamics, with the legend given in part a. (a) Cell potential. (b) Electrolyte Li-ions concentration. (c) Solid phase Li-ions concentration. (d) Electrolyte potential. (e) Temperature.

Figure 7. Comparison of the three different solid-phase diffusion equations implemented in LIONSIMBA. (a) 1C rate comparison. (b) 2C rate comparison. (c) 5C rate comparison. (d) 10C rate comparison.

In the third scenario, the framework is used to simulate a hybrid charge-discharge cycle, emulating the throttle of a HEV. During braking, the battery is charged. Table V resumes the configuration of the car throttle during simulations. In Fig. 10 it is possible to analyze the response of a single cell inside an HEV pack under a hybrid charge-discharge cycle. In this case, the effects of temperature among the different cells have been neglected. The solid potential behavior is primarily due to the different applied C rates, with discontinuous changes producing voltage drops. Different slopes of the voltage curve are related to the different C rates applied. Temperature rise is...
recorded in the first 50 seconds of simulations, which are followed by a slight decrease of the temperature mainly due to the exchange of heat with the surrounding environment \( h = 1 \text{ W/(m}^2\text{ K)} \) and due to the lower current density applied. At around 250 s, the temperature starts to increase due to the 1C rate applied during moderate speed; the high slope of increase at around 410 s is due to the higher value of the discharge current which during an overtake reaches the value of 2C. Returning to moderate speed makes the temperature slope more

Figure 8. 1C discharge cycle run under different heat exchange parameters: blue line \( h = 0.01\text{ W/(m}^2\text{ K)} \), dashed orange line \( h = 1 \text{ W/(m}^2\text{ K)} \) and dot-dashed yellow line \( h = 100 \text{ W/(m}^2\text{ K)} \).

Figure 9. Full discharge cycle run under different C rates: 2C (dot-dashed yellow), 1C (dashed orange line), and 0.5C (blue line).

Figure 10. Hybrid charging-discharging cycle.

Table V. Throttle configuration for hybrid charging-discharging simulation.

| Time (s) | C rate | Description   |
|---------|--------|---------------|
| 0–50    | –1 C   | Moderate speed|
| 50–60   | 0.5 C  | Charge        |
| 60–210  | –0.5C  | Normal speed  |
| 210–410 | –1C    | Moderate speed|
| 410–415 | –2C    | Overtaking    |
| 415–615 | –1C    | Moderate speed|
| 615–620 | 0.5C   | Charge        |
gentle. During the last 10 seconds, temperature decreases due to the significant change in applied current and due to dissipation of heat with surrounding ambient. A sketch of the code used for this simulation is presented in Algorithm 1.

**Algorithm 1** Car cycling example code.

**Input setup:**
1: \( I = \{-29.5, 14.75, -14.75, -29.5, -58, -29.5, 14.75\} \)
\( \triangleright \) Simulation current densities
2: \( \text{time} = \{50, 10, 150, 200, 5, 200\} \)
\( \triangleright \) Duration of each element of \( I_{\text{applied}} \) (in seconds).
3: \( t_0 = 0; \)
\( \triangleright \) Init all the useful variables
4: \( t_f = 0; \)
5: \( \text{initialStates.Y} = []; \)
6: \( \text{initialStates.YP} = []; \)
7: \( \text{Phis_tot} = []; \)
8: \( t_{\text{tot}} = []; \)
9: \( T_{\text{tot}} = []; \)

**Core script:**
10: for \( i = 1:\text{length}(I) \) do
11: \( t_f = t_f + \text{time}(i); \)
12: \( \text{results} = \text{startSimulation}(t_0,t_f,\text{initialStates},I(i),[]); \)
13: \( \text{Phis_tot} = [\text{Phis_tot},\text{results.original.Phis}]; \)
\( \triangleright \) Concatenate results
14: \( \text{T_tot} = [\text{T_tot},\text{results.original.Temperature}]; \)
15: \( t_0 = \text{time}(i); \)
16: \( \text{initialStates} = \text{results.initialStates}; \)
\( \triangleright \) Update initial states for the next simulation
17: end for

The simulation of the application of an ABMS is shown in Fig. 11. In this particular simulation, a model predictive control algorithm is adopted to drive the State of Charge (SOC) of the battery to a given value, while accounting for input and output constraints. The initial SOC was around 20% and its reference value was set to 85%. According to LIONSIMBA, the estimation of the SOC can be easily simulated by defining a custom function. In this particular scenario, the SOC has been computed as

\[
\text{SOC}(t) = \frac{1}{l_n s_{\text{max}}} \int_{s_0}^{s_{\text{max}}} c^{\text{ref}}(x,t) \, dx
\]

**Algorithm 2** High level control code.

**Init script:**
1: \( t_0 = 0; \)
2: \( t_f = dt; \) \( \triangleright \) Simulations are run over a sampling time periods
3: \( \text{initialStates.Y} = []; \)
4: \( \text{initialStates.YP} = []; \)
5: \( \text{Condition} = 1; \)

**Core script:**
6: while \( \text{Condition} \) do
7: \( I = \text{ComputeControlLaw}(\text{initialStates}); \)
8: \( \text{results} = \text{startSimulation}(t_0,t_f,\text{initialStates},I,[ ]); \)
9: \( [...] \)
\( \triangleright \) Elaborate and concatenate the results and update the time indexes
10: \( \text{initialStates} = \text{results.initialStates}; \)
\( \triangleright \) Update initial states for the next simulation
11: if SOC reached reference value then
12: \( \text{Condition} = 0 \)
13: \( \text{end if} \)
14: end while

Figure 11. ABMS control: an MPC algorithm is used to drive the charge of the battery from 20% to 85% while considering voltage, temperature, and current constraints. The temperature maximum bound was set to 313.5 K, with the voltage set to 4.2 V. The ABMS applies a current density which is almost fixed at 1C value for the entire simulation, while starting to drop as the SOC approaches its final value. The SOC raises almost linearly during the first 2500 s, then plateauing according to the current drop, in order to approach smoothly the final stage of charge. Fig. 12 shows that, according to the different charging stages, the electrolyte concentration diffuses in different ways. Starting from \( c_{\text{init}} = 1000 \, \text{mol/m}^3 \), the input current induces a drop in electrolyte concentration within the battery sections due to the diffusion of ions from the cathode to...
Table VI. Comparison of different approximation methods for the diffusion in the solid particles. Root Mean Square Error (RMSE) and the Normalized Time Index (NTI) are shown.

| C rate | RMSE   | NTI |
|--------|--------|-----|
|        |        |     |
| 1C     | 0.082% | 20% |
| 2C     | 0.25%  | 18% |
| 5C     | 1.6%   | 22% |
| 10C    | 6.6%   | 24% |

![Figure 12. ABMS control – Electrolyte concentration: The behavior of the first and last volume of each section of the battery is depicted, where the continuous lines belong to the cathode, the dashed lines to the separator and the dotted lines to the anode.](image)

Figure 12. ABMS control – Electrolyte concentration: The behavior of the first and last volume of each section of the battery is depicted, where the continuous lines belong to the cathode, the dashed lines to the separator and the dotted lines to the anode.

![Figure 13. Full discharge cycle in an isothermal environment: blue line 0.5C, dashed orange line 1C, and dot-dashed yellow line 2C.](image)

Figure 13. Full discharge cycle in an isothermal environment: blue line 0.5C, dashed orange line 1C, and dot-dashed yellow line 2C.

the anode. Approaching the final stage of charging, the concentration starts to converge back to the initial value of 1000 mol/m³ and, around 5500 s, reaches the steady value. This behavior emphasizes the property of the FVM to conserve properties within numerical roundoff. Algorithm 2 provides a high-level description of how to implement a closed-loop controller in LIONSIMBA.

In Fig. 13, simulations have been run disabling the thermal dynamics leading to an isothermal environment. This particular configuration can be exploited in order to assess the influence of different constant temperatures at which the battery can operate.

All the results of the proposed simulations can be reproduced by running the example scripts available with LIONSIMBA. Table VII shows the times required by the simulator to simulate the different scenarios, which are all under 100 s.

**Battery pack of series-connected cells.**—The results of a battery pack simulation are shown in Figs. 14 and 15. To emphasize the ability to independently parametrize each cell, in this scenario the SOC of cell #1 is set to the 95% of its initial value while the thickness of the cathode of cell #2 is doubled with respect to its initial value. All the other parameters are the same for the three cells. The time responses of the output voltage of the overall pack and the voltage of each cell are plotted in Fig. 14. The starting voltage of the pack is around 12.1 V and decreases subjected to a 1C discharge current. In 3346 s, the pack is completely discharged due to cell #1 first reaching the cutoff voltage (set to 2.5 V). The lowered starting SOC determined this behavior. The electrolyte and solid-phase surface concentrations as well as the electrolyte potentials are compared for the three cells in Fig. 15. Cell #2 has a significantly different behavior mainly due to the presence of a cathode with a thickness two times that of the other two cells. This variation has effects over the output voltages, as shown in Fig. 14. Besides cell #1 which is starting from a different SOC value, the different behaviors of $V(t)$ between cell #2 and cell #3 are driven by the thickness variation.

**Conclusions**

This work describes a detailed procedure for the numerical implementation of the P2D model. Two published approximate models for the solid-phase diffusion are also implemented, in which the pseudo-second dimension is removed to reduce the computational complexity. The treatment of boundary conditions is addressed with particular attention to the interface conditions across the different sections of the battery. Following the procedures and rules outlined in Numerical implementation section, the reader can implement his/her own version of the model in different programming languages. Moreover, a freely available Matlab framework LIONSIMBA is provided that is suitable for battery design, simulation, and control. The framework is extended to account for different solid-phase diffusion models to meet required accuracy. The simulations demonstrate high numerical stability for different operating scenarios. The effectiveness and reliability of LIONSIMBA is verified considering a heterogeneous sequence of applied current coming from an HEV and through the assessment of an ABMS strategy, in particular, the model predictive control of state of charge.

A battery pack composed of series-connected cells can be simulated by considering several independent cells with their own parameters. Due to its integration with the Matlab environment, the framework facilitates the development and test of different algorithms such as control algorithms, identification procedures, or optimization of manufacturing parameters. The computational times of LIONSIMBA are compared to DUALFOIL and COMSOL in Table VIII. For each code, average simulation times are reported for repeated simulations of a 1C discharging cycle in isothermal conditions. LIONSIMBA and DUALFOIL have very similar computation times for all discretizations, with COMSOL being significantly slower at lower discretizations. For the same numerical algorithm, an implementation in a compiled language such as Fortran is inherently much faster than an interpreted language such as Matlab, indicating that the underlying numerical algorithm used by LIONSIMBA is more suitable for battery design, simulation, and control. The framework facilitates the development and test of different algorithms such as control algorithms, identification procedures, or optimization of manufacturing parameters. The computational times of LIONSIMBA are compared to DUALFOIL and COMSOL in Table VIII. For each code, average simulation times are reported for repeated simulations of a 1C discharging cycle in isothermal conditions. LIONSIMBA and DUALFOIL have very similar computation times for all discretizations, with COMSOL being significantly slower at lower discretizations. For the same numerical algorithm, an implementation in a compiled language such as Fortran is inherently much faster than an interpreted language such as Matlab, indicating that the underlying numerical algorithm used by LIONSIMBA is more

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Table VII. Timing comparisons of different simulation scenarios.

| C rate | h value | Simulation Duration | Effective Simulation Time |
|--------|---------|---------------------|--------------------------|
| 1C     | 0.01    | 3523 s              | 72 s                     |
| 1C     | 1       | 3523 s              | 81 s                     |
| 1C     | 100     | 3523 s              | 77 s                     |
| 0.5C   | 1       | 7050 s              | 56 s                     |
| 2C     | 1       | 1522 s              | 85 s                     |

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efficient but the higher efficiency is offset by Matlab being an interpreted language: the two effects approximately cancel so that the overall simulation times for LIONSIMBA and DUALFOIL were very similar.

The results in this article demonstrate the promise of the proposed framework as a reliable, efficient, and freely available Matlab-based software for the P2D model simulation. Further developments such as code optimization and distribution of compiled versions can only improve the current performance. Moreover, as the proposed simulations were written in standard serial mode, the computation time could be reduced by at least a factor of ten by using a multicore CPU using parallel DAE solvers. Modern versions of Matlab have easy-to-implement built-in options for distributing calculations among multiple cores on a single CPU, and among multiple CPUs.

| # of discrete nodes | 10  | 20  | 30  | 40  | 50  |
|--------------------|-----|-----|-----|-----|-----|
| COMSOL             | 96 s| 114 s| 143 s| 189 s| 244 s|
| DUALFOIL           | 28 s| 57 s| 97 s| 137 s| 185 s|
| LIONSIMBA          | 28 s| 69 s| 105 s| 134 s| 223 s|

**List of Symbols**

- \( c_s(x, t) \) Electrolyte salt concentration [mol/m\(^3\)]
- \( c_{sw}^a(x, t) \) Solid-phase average concentration [mol/m\(^3\)]
- \( c^*_{sw}(x, t) \) Solid-phase surface concentration [mol/m\(^3\)]
- \( D_{eff} \) Effective electrolyte diffusion coefficient [m\(^2\)/s]
- \( D_{eff}^{sp} \) Effective solid-phase diffusion coefficient [m\(^2\)/s]
- \( h \) Heat exchange coefficient [W/(m\(^2\)K)]
- \( I_{app}(t) \) Applied current density [A/m\(^2\)]
- \( j(x, t) \) Ionic flux [mol/(m\(^2\)s)]
- \( k_{eff} \) Effective reaction rate [m\(^2\)s/(mol\(^0.5\))]\(^{.5}\]
- \( Q_{ohm} \) Ohmic heat source term [W/m\(^3\)]
- \( Q_{rev} \) Reversible heat source term [W/m\(^3\)]
- \( Q_{rxn} \) Reaction heat source term [W/m\(^3\)]
- \( T(x, t) \) Temperature [K]
- \( U_{ref} \) Open Circuit Voltage [V]
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