Original Article

Experimental studies and thermodynamic assessment of the Ba-Mo-O system by the CALPHAD method

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\textbf{ABSTRACT}

Thermodynamic measurements on BaMoO\textsubscript{4}, BaMoO\textsubscript{3} and BaMo\textsubscript{3}O\textsubscript{10} are reported, that served as input for the development of a thermodynamic model of the Ba-Mo-O system using the CALPHAD methodology. The valence states of molybdenum in BaMoO\textsubscript{4} and BaMoO\textsubscript{3} were confirmed to be VI and IV, respectively, from X-ray Absorption Near Edge Structure Spectroscopy measurements at the Mo K-edge. The heat capacity at low temperatures of these compounds was obtained from thermal-relaxation calorimetry. Phase equilibrium data in the BaMoO\textsubscript{4}-MoO\textsubscript{3} section were also measured, and the transition enthalpy associated with the peritectic decomposition of BaMo\textsubscript{3}O\textsubscript{10} was determined using Differential Scanning Calorimetry. The developed thermodynamic model used the compound energy formalism for intermediate compounds, and an ionic two-sublattice model for the liquid phase. The optimized Gibbs energies were assessed with respect to the known thermodynamic and phase equilibrium data. A good agreement is generally obtained, but a number of ill-defined data were also identified.

\section{1. Introduction}

The chemistry of fission product (FP) elements in irradiated nuclear fuel, and more particularly of volatile and semi-volatile elements, is of paramount importance, as FPs are the main source for the radiological consequences of a severe accident (SA) with release to the environment. Barium and molybdenum are generated with a high fission yield (~11\% and ~25\%, respectively [1]) during irradiation of the UO\textsubscript{2} or (U,Pu)O\textsubscript{2} ceramic fuel used in Light Water Reactors (LWRs), and represent key elements for the evaluation of the source term. They are classified as semi-volatile fission products, which implies that their release kinetics are dependent on the redox conditions of the surrounding environment, and are determined by the evaporation of the chemical compounds formed in the irradiated fuel [1]. The association of barium with molybdenum in the nuclear fuel in the form of oxide precipitates at grain boundaries was recently confirmed from the FPT2 test of the PHEBUS Fission Product Program [2]. However, the exact chemical form of barium and molybdenum in the fuel is rather complex, and changes with time in relation with the burnup, temperature and oxygen potential conditions. The severe accident at the Fukushima-Daiichi Nuclear Power Station (FDNPS) has initiated a renewed interest in the behaviour of radionuclides with a potential long-term radiological impact such as those of Ba, Sr, U, Pu, and minor actinides. To respond to this necessity, the TCOFF project (Thermodynamic Characterization of Fuel Debris and Fission Products Based on Scenario Analysis of Severe Accident Progression at Fukushima-Daiichi Nuclear Power Station) was launched in 2017 under the auspices of the OECD/NEA [3], and the experimental study of the Ba-Mo-O system as outlined in this work constitutes a contribution to that initiative.

The solubility of barium in the (U,Pu)O\textsubscript{2} fuel matrix is very low due to its large ionic radius. At low oxygen potentials, it is found in the so-called grey phase of general formula (Ba,Sr,Cs)(Zr,U,Mo,RE)O\textsubscript{3} (RE = rare earths) with perovskite structure [4–8], while it is stable in the Ba(Mo,U)O\textsubscript{3} scheelite phase at high oxygen potentials [5,7–9]. For a thorough safety assessment of nuclear fuel behaviour under operation and in accidental conditions, a complete thermodynamic description of the multi-component system Ba-Sr-Cs-Zr-U-Pu-Mo-RE-O is therefore necessary. A number of thermodynamic and thermophysical data on the
oxide phases formed in this system have been reported, but the knowledge is still limited. Nevertheless, a thermodynamic model has been developed using a CALPHAD (CALculation of PHAse Diagram) approach within the TAF-ID project (Thermodynamics of Advanced Fuels – International Database) of the OECD/NEA [10]. More specifically, thermodynamic assessments of the Ba-Zr-O, Ba-U-O, Cs-Mo-O, Cs-U-O, Cs-Zr-O and Sr-Zr-O sub-systems have been made. The present work reports, next to the experimental studies, a CALPHAD model for the Ba-Mo-O system. It constitutes another building unit of the multi-element database of the TAF-ID. It also contributes to improving our knowledge of the still poorly understood thermochemistry of Ba and Mo fission product phases under severe accident conditions, which is key for a more reliable assessment of the source term in the SA codes.

The interest in the phase relationships in the Ba-Mo-O system does not limit itself to nuclear energy applications, however. Molybdate materials have received quite some attention in recent years because of their interesting properties for applications in photoluminescence [11–13], solid-state lasers [14,15], photocatalysts [16,17], gas sensing [18], microwave [19,20], or thermoelectric applications [21]. BaMoO$_4$ for instance shows photoluminescence [12,22] and can be used as host material for lanthanide cations (activated phosphors) [22].

In this work, we report a literature review of the thermodynamic and phase diagram data available on the Ba-Mo-O system, the synthesis and structural characterization of BaMoO$_4$, BaMoO$_3$, and BaMo$_3$O$_{10}$, the measurement of the low-temperature heat capacity of the former two phases, the determination of the transition enthalpy associated with the peritectic decomposition of BaMo$_3$O$_{10}$, and phase diagram measurements in the BaMoO$_4$-MoO$_3$ pseudo-binary section. Using these data as input, a CALPHAD model of the Ba-Mo-O system is then developed, that is compatible with the current descriptions and formalisms used in the TAF-ID database [10].

2. Review of literature data on the Ba-Mo-O system

2.1. Constituting binary sub-systems

A thermodynamic assessment of the three binary sub-systems, i.e. Ba-O, Ba-Mo, and Mo-O, is already available as part of the TAF-ID database of the OECD/NEA [10]. The latter were used as basis for the present assessment, and this work therefore focuses on the phase equilibria in the ternary system. The corresponding binary phase diagrams are shown in Fig. 1a–c.

2.1.1. Binary Ba-Mo system

Ba and Mo are stable under the same crystalline phase, i.e. the bcc phase. However, their chemical and physical properties strongly differ, leading to a very small reciprocal solubility. The thermodynamic assessment of the Ba-Mo system was made in the framework of the Fuelbase project (2011) [23]. The corresponding parameters are listed in Tables C.1 and C.2. Due to the scarcity of data on this system, identical interaction parameters of the elements have been assumed in the liquid phase and in the bcc phase. This choice, made in the absence of

Fig. 1. Calculated binary phase diagrams of the (a) Ba-O (b) Mo-O and (c) Ba-Mo sub-systems using the TAF-ID models [10].
sufficient information, presumes similar first nearest neighbour
coordination spheres in the liquid and bcc phases. The values of those
interaction parameters have been derived in order to agree with the
maximum solubility in the Ba liquid phase proposed by Massalski [24].
The formalisms used are described in Section 4.

2.1.2. Binary Ba-O system

The description initially present in the TAF-ID database for Ba-O was
inherited from the Fuelbase project [23], based on the work by Zim-
mermann et al. [25]. The current description was built starting from the
later assessment by Zhou et al. [26], that allows to better describe the
heat capacity data for BaO2, following measurements reported subse-
tively to the work of Zimmermann et al. The description of the condensed phases BaO and BaO2 by Zhou et al. have been modified to be consistent with the current models in the TAF-ID, however. The O2−
species has not been considered in the model for the liquid phase, but
rather has been replaced by a neutral O species. The stability of the
liquid phase for compositions richer in oxygen than BaO has not been
forced as in the original description, since no experimental information
was available on this issue. The solid phases have been considered as
stoichiometric. The BaO2 homogeneity range modelled by Zimmermann
et al. and by Zhou et al. is in fact considered too large according to [27]
and [28]. This simplification implied a slight modification of the BaO2
stoichiometric compound description, so as to maintain the agreement
with the oxygen pressure data over the BaO-BaO2 region. The descrip-
tion of the gaseous species in this system, and in particular of the binary
species BaO and Ba2O, has been extracted from the substance SGTE (Scientific Group Thermodata Europe) database. The thermodynamic function for the gaseous Ba2O2 species was taken from the work of Zimmermann et al.

The optimized parameters are listed in Tables B.1 and B.2, and a
comparison with the experimental data is shown in the Appendix (Fig.
17B.1), (Fig. 18B.2), (Fig. 19B.3), (Fig. 20B.4), (Fig. 21B.5), (Fig. 22B.6). The formalisms used for the different phases are described

| Table 1 |
| --- |
| Structural data on the Ba-Mo-O system. The phases for which no atomic posi-
tions have been reported are indicated with an *. |

| Phase | Ox. state | Symmetry | Space group | Lattice parameters | Ref. |
| --- | --- | --- | --- | --- | --- |
| BaMoO4 | +6 | Tetragonal | 4/-a (88) | a = 0.5571(3) nm | [31] |
| Ba2MoO6 | +6 | Orthorhombic | Pnma (62) | a = 0.7412 nm | [32] |
| BaMoO3 | +6 | Cubic | Fm3m (225) | a = 0.8600 nm | [33] |
| BaMoO3 | +6 | Monoclinic | P21 (4) | a = 1.4695(2) nm | [34] |
| BaMoO3 | +6 | Unknown | [ref] | | |
| BaMoO3 | +3 | Orthorhombic | Pnma (62) | a = 1.0154 nm | [37] |
| BaMoO3 | +2.78 | Monoclinic | Pca2 (14) | a = 0.9939(2) nm | [38] |
| BaMoO3 | +2.69 | Orthorhombic | Pnma (55) | a = 0.9517(1) nm | [39] |

| Table 2 |
| --- |
| Experimental measurements of the enthalpy of formation of BaMoO4(cr). |

| Reference | Method | ΔfHm°(298.15 K) (kJ mol⁻¹) |
| --- | --- | --- |
| Tamman and Westerhold [40], [42] | Indirect (Reaction) | Ba₂O + MoO₃ → BaMoO₄ at 560 K | (1540 ± 13) |
| Lavrent’ev et al. [41, 42] | Indirect (Oxygen potential of phase mixture of Ba₂O, Mo, BaMoO₃) | | (1586.8 ± 10.3) |
| Renakshina [43, 47] | Indirect (EMF measurements) | | (1501.4 ± 10.7) |
| O’Hare [44] | Direct (solution calorimetry) | | (1545.6 ± 1.9) |
| Shukla et al. [45] | Direct (solution calorimetry) | | (1574.8 ± 3.6) |
| Singh et al. [46] | Indirect (EMF measurements) – second law | | (1557.5 ± 3.6) |
| Cordfunke and Konings [42] | Review | | (1536.8 ± 5.2) |

in Section 4.

2.1.3. Binary Mo-O system

The Mo-O system is quite complex with five binary oxide compounds
reported, i.e. MoO2, Mo2O3, Mo2O5, Mo3O7, Mo5O23, also treated as
stoichiometric. The optimized parameters for the Mo-O system are re-
ported in the work of Corcoran et al. [29]. The parameters for the liquid
phase have been later updated in the PhD thesis of Kauric [30] to pre-
vent the occurrence of a miscibility gap at very high MoO2 content in
terary systems such as Cs-Mo-O or Na-Mo-O.

2.2. Structural data on the ternary Ba-Mo-O phases

The Ba-Mo-O system is rather complex, with a number of ternary
phases reported: BaMoO4, Ba2MoO6, Ba3MoO8, BaMoO10, Ba3MoO7,
Ba3MoO11, Ba3MoO12, Ba4MoO16, Ba5MoO18, and Ba6(MoO3)6. The reported crystal structures for each
phase are listed in Table 1. The availability of the atomic positions for
the crystal structures is also indicated. In the present thermodynamic
model, only the hexavalent phases BaMoO4, Ba2MoO6, Ba3MoO8,
BaMoO10, Ba5MoO18, and pentavalent phase BaMoO5 are considered, for
which sufficient structural and thermodynamic information is available, and whose existence are most probable. Trivalent BaMoO10 and
the reported mixed valence state solid phases Ba7/8Mo5O16, Ba3MoO11,
Ba3MoO18 and Ba3(MoO2)6 (i.e. Mo at the valence states +3, +2.78, +2.69, respectively), were not included due to the lack of ther-
modinamic data, and because of the general scarcity of experimental
information on the phase diagram equilibria in the regions of compo-
sition where such phases should be stable.

2.3. Thermodynamic data

Thermodynamic data in the literature are only available to the best of
our knowledge on BaMoO4, BaMoO3, and BaMoO2.

2.3.1. BaMoO4

The enthalpy of formation of BaMoO4 was determined by various
authors using indirect measurement techniques and solution calorim-
etry. The values reported are summarized in Table 2. Tamman and
Westerhold [40] derived the enthalpy of formation from the measure-
ment of the enthalpy of reaction between barium oxide and molybd-
enum trioxide at T = 560 K:

\[ \text{BaO(cr) + MoO}_3(cr) = \text{BaMoO}_4(cr) \]
K [41]. However, the review of Cordfunke and Konings [42] suggested that equilibrium conditions were not reached during those experiments, explaining the discrepancy with other reported values. Rezukhina performed emf (electromotive force) measurements and studied reaction (Eq. (1)) [43]. Kubaschewski reviewed those measurements and derived \( \Delta H_\text{m} = -(208.4 \pm 10.5) \text{ kJ} \cdot \text{mol}^{-1} \) for reaction (Eq. (1)). Combining with the enthalpies of formation of BaO and MoO\(_3\) recommended by [42], we obtain an enthalpy of formation or BaMoO\(_4\) of \((1501.4 \pm 10.7) \text{ kJ} \cdot \text{mol}^{-1}\). O’Hare [44] studied the precipitation of BaMoO\(_4\) from an ammoniacal solution of BaCl\(_2\), and derived the enthalpy of formation based on the reaction:

\[
C_{60}\text{MoO}_4(\text{cr}) + Ba\text{Cl}_2(\text{cr}) = Ba\text{MoO}_4(\text{cr}) + 2C\text{aCl}(\text{aq})
\]  

Shukla et al. [45] also studied the precipitation of BaMoO\(_4\) (cr), but from an ammoniacal solution of Ba(NO\(_3\))\(_2\), and derived the enthalpy of formation based on the reaction:

\[
Na_2\text{MoO}_4(\text{cr}) + Ba(\text{NO}_3)_2(\text{cr}) = Ba\text{MoO}_4(\text{cr}) + 2NaNO_3(\text{aq})
\]  

Singh et al. measured the standard molar Gibbs energy of formation of BaMoO\(_4\) by measuring the emf of the cell (Pt/BaMoO\(_4\)+BaMoO\(_4\)/CSZ/ air (p(O\(_2\)) = 21.21 kPa, CSZ = 15 mol% CaO stabilized zirconia) in the temperature range \(T = (1091–1309) \text{ K}\), and derived the standard enthalpy of formation by second and third law analyses of the data [46]. A more detailed description of the individual measurements can be found in [42], [45] and [46].

The values obtained by direct solution calorimetry measurements are preferred in this work. They are in good agreement and based on two different reaction schemes. Here we recommend the weighted\(^1\) average of the aforementioned two values, i.e. \( \Delta_f H_\text{m}^\circ (\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = -(1546.1 \pm 1.7) \text{ kJ} \cdot \text{mol}^{-1} \).

The low-temperature heat capacity was measured from 2.02 to 297.19 K using the thermal-relaxation method [49]. The heat capacity and the standard entropy at 298.15 K were determined in that study as \( C_{p,m}(\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = 118.47 \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \) and \( S_{m}^\circ (\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = (152.69 \pm 1.53) \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \), respectively. The latter data were re-measured in this work using again the thermal-relaxation technique in the temperature range 2.1 to 293.2 K, which yielded higher values, namely \( C_{p,m}(\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = (126.4 \pm 3.8) \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \) and \( S_{m}^\circ (\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = (160.4 \pm 3.8) \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \), respectively. The latter data are preferred and retained for the thermodynamic model, for the reasons detailed in Section 5.3.

High temperature enthalpy increment measurements were reported using drop calorimetry in the temperature range \(T = (986–1732) \text{ K}\) by Saha et al. [50], and in the temperature range \(T = (299–1020) \text{ K}\) by Singh et al. [51]. From the data of Saha et al. [50], and constraining the fit to \( C_{p,m}(\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = 122.2 \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \), Cordfunke and Konings [42] derived the equation:

\[
\frac{H_\text{m}(T) - H_\text{m}(298.15 \text{ K})}{J \cdot \text{mol}^{-1}} = 91.9766\left(\frac{T}{K}\right) + 35.533410^{-3}\left(\frac{T}{K}\right)^2 - 7.8535210^{-5}\left(\frac{T}{K}\right)^3 - 27947.5.
\]

Singh et al. [51] reported the following equation by constraining to the same heat capacity value at 298.15 K, and combining their data with that of Saha et al. [50]:

\[
\frac{H_\text{m}(T) - H_\text{m}(298.15 \text{ K})}{J \cdot \text{mol}^{-1}} = 138.510\left(\frac{T}{K}\right) + 12.50910^{-3}\left(\frac{T}{K}\right)^2 + 21.10110^{-5}\left(\frac{T}{K}\right)^3 - 49486.
\]

More recently, Saha et al. [52] reported direct measurements of the heat capacity of BaMoO\(_4\) in the temperature range \(T = (140–870) \text{ K}\) using Differential Scanning Calorimetry (DSC) and the step method. In this work, we have refitted the data of [50] and [51] using the constraints \( H_\text{m}(T) - H_\text{m}(298.15 \text{ K}) = 0 \) at 298.15 K and \( C_{p,m}(\text{BaMoO}_4, \text{cr}, 298.15 \text{ K}) = 126.4 \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \), which yielded:

\[
C_{p,m}(\text{BaMoO}_4, \text{cr}, T) / J \cdot \text{K}^{-1} \cdot \text{mol}^{-1} = 121.7475 + 3.918 \times 10^{-2}(T / K) - 6.248339 \times 10^7 \left(\frac{T}{K}\right)^{-2}
\]

The latter equation is used for the thermodynamic model and compared in Figs. 2a and 7a to the experimental data, and selected equations by Cordfunke and Konings [42] and Singh et al. [51].

BaMoO\(_4\) was reported to melt congruently at \(T = 1737 \text{ K}\) by Dash et al. [53], but the original source is not cited. It was moreover reported to melt at \(T = 1723 \text{ K}\) in the Handbook of Inorganic Compounds [54] based on the review by Knacke et al. [55]. A previous study by Ustinov et al. [56] reported a melting temperature at \(T = 1273 \text{ K}\), but this is highly

![Fig. 2. (a) Enthalpy increments of BaMoO\(_4\) and fit to the data. (b) Heat capacity of BaMoO\(_4\) and fit to the data.](image-url)

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1 The weighted average was calculated as recommended by Guillaumont et al. for independent source data [48].
unlikely for the reasons detailed in Section 2.4. The enthalpy of fusion of BaMoO$_3$ was not reported in the literature. By comparison with other isomorphous alkaline-earth molybdates and tungstates (CaMoO$_4$ and BaWO$_4$), and assuming a constant fusion entropy for the series of isomorphic compounds, one can expect this fusion enthalpy to be in the order of 94 to 102 kJ·mol$^{-1}$. The fusion temperature and fusion enthalpy of CaMoO$_4$ were reported as (1750 ± 4) K and (102 ± 6) kJ·mol$^{-1}$ [57], respectively. The same data for BaWO$_4$ were found as 1775.1 K and 96.9 kJ·mol$^{-1}$ [58], respectively.

2.3.2. BaMoO$_3$

The enthalpy of formation of BaMoO$_3$ was determined using the emf technique [59], oxygen bomb calorimetry [60] and Knudsen mass-loss method [61]. The values reported in the literature are summarized in Table 3. The values obtained by Dash et al. [61] by second and third laws are in good agreement, but about 35 kJ·mol$^{-1}$ lower than the results in the other studies. Here we prefer to retain the recommendation of Cordfunke and Konings [42] based on the average values of Zharkova et al. [60] and Rezukhina and Levitskii [59]:

$$
\Delta H_f^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) = -(1227.0 ± 15.0) \text{kJ·mol}^{-1}
$$

There are no low-temperature heat capacity data available for this compound in the literature. Cordfunke and Konings [42] have estimated the heat capacity and entropy based on a comparison with BaZrO$_3$: $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) \sim (101.7 ± 5.0) \text{J·K}^{-1}·\text{mol}^{-1}$ and $S_f^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) \sim (43.6 ± 5.0) \text{J·K}^{-1}·\text{mol}^{-1}$, respectively. The enthalpy values reported by Rezukhina [43] and Rezukhina and Levitskii [59] on emf studies are 119.43 and 92.04 J·K$^{-1}$·mol$^{-1}$, respectively. Dash et al. estimated 82.28 J·K$^{-1}$·mol$^{-1}$ based on the Knudsen effusion mass-loss measurements [61], while Yokokawa et al. [62] estimated 125.7 J·K$^{-1}$·mol$^{-1}$. It is clear that there is a large dispersion of reported values, and low-temperature heat capacity measurements are reported in this work to solve these discrepancies (see Section 5.3).

The obtained data, selected for the present thermodynamic model, are $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) = (103.4 ± 3.1) \text{J·K}^{-1}·\text{mol}^{-1}$ and $S_f^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) = (111.3 ± 2.8) \text{J·K}^{-1}·\text{mol}^{-1}$, respectively.

Agarwal et al. [63] reported enthalpy increment measurements in the temperature range $T$ = (477.8–1010.8) K using a high temperature Calvet calorimeter. The fitting of this data [63] using the constraint $[H_f(T) - H_f(298.15 \text{K})]/J·\text{mol}^{-1} = 0$ at 298.15 K yielded: $[H_f(T) - H_f(298.15 \text{K})]/J·\text{mol}^{-1} = -31427 + 97.977(T/\text{K}) + 0.0275(T/\text{K})^2$, and by derivation $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, T)/\text{J·K}^{-1}·\text{mol}^{-1} = 97.977 + 0.0550(T/\text{K})$. The extrapolation of the latter equation to 298.15 K yields $114.5 \text{J·K}^{-1}·\text{mol}^{-1}$, which is about 11.9 J·K$^{-1}$·mol$^{-1}$ higher than measured in this work. Yamanaka et al. [64] reported heat capacity measurements in the temperature range $T$ = (300 to 1200) K using DSC with triple cells in a high purity argon atmosphere. The heat capacity was determined by both enthalpy and scanning methods at individual temperatures. Unfortunately the numerical values have not been listed in their paper. Only the fitted equation is given: $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, T)/\text{J·K}^{-1}·\text{mol}^{-1} = 131.8 + 9.36 (10^3/T/\text{K})^{-2}$. The extrapolation of the latter equation to 298.15 K yields 85.88 J·K$^{-1}$·mol$^{-1}$, which is much lower than expected, however. Sahu et al. [65] also reported heat capacity measurements in the temperature range $T$ = (300 to 870) K using DSC and the step method. Here again, the numerical values have not been listed. Only the fitted equation is given: $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, T)/\text{J·K}^{-1}·\text{mol}^{-1} = 123.25 + 30 (10^3/T/\text{K})^{-2}$. The extrapolation of the latter equation to 298.15 K yields $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K})/\text{J·K}^{-1}·\text{mol}^{-1} = 101.2 \text{J·K}^{-1}·\text{mol}^{-1}$, in good agreement with the data measured herein. The measurements of Agarwal et al. [63] and Sahu et al. [65] are compared in Fig. 3a and b. Both equations yield similar values above $T = 450$ K, but differ largely on the extrapolation down to room temperature.

In this work, we have refitted the data of Agarwal et al. [63] and Sahu et al. [65] using the constraints $[H_f(T) - H_f(298.15 \text{K})]/J·\text{mol}^{-1} = 0$ at 298.15 K and $C_p^{\circ}(\text{BaMoO}_3, \text{cr}, 298.15 \text{K}) = 103.4 \text{J·K}^{-1}·\text{mol}^{-1}$ (obtained in this work), which yielded:

$$
C_p^{\circ}(\text{BaMoO}_3, \text{cr}, T)/\text{J·K}^{-1}·\text{mol}^{-1} = 128.174 + 2.31710^{-2}(T/\text{K})^{-2} - 2.81632710^{6}(T/\text{K})^{-2}
$$

The latter equation was implemented in our thermodynamic model.

Paschoal et al. [66] reported the decomposition of BaMoO$_3$ at $T = 1653$ K into BaMoO$_4$ (cr), Mo(cr) and BaO(g) according to Differential Thermal Analysis (DTA) measurements under argon and helium atmospheres, i.e. following the equilibrium reaction $3\text{BaMoO}_3(\text{cr}) = 2\text{BaMoO}_4(\text{cr}) + \text{Mo(cr)} + \text{BaO(g)}$. More recently, Yamanaka et al. [64] measured the melting temperature of BaMoO$_3$ using a thermal arrest method under a reducing atmosphere and reported $T_m(\text{BaMoO}_3) = 1791$ K. No enthalpy of fusion data are available to this date on this compound.

2.3.3. BaMo$_2$O$_7$

Singh et al. [67] measured the Gibbs energy of formation of BaMo$_2$O$_7$ in the temperature range $T$ = (828–985) K using an emf cell: ([−] Pt/BaMo$_2$O$_7$/BaMoO$_4$ + Mo$_2$O$_7$/CSZ/air (p(O$_2$(g))=21.21 kPa)/Pt([−]) (CSZ = 0.15% calcia-stabilized zirconia)). From a second law analysis of

$$\Delta G^{\circ}(\text{BaMo}_2\text{O}_7, \text{cr}, 298.15 \text{K}) = -112.7 \text{kJ·mol}^{-1}$$
their data, the authors derived the enthalpy of formation at 298.15 K as 
\[ \Delta_H^0(\text{BaMo}_2\text{O}_7, cr, 298.15 \text{ K}) = -(2291.2 \pm 2.5) \text{ kJ \cdot mol}^{-1} \]. The heat capacity was not measured to this date.

2.4. Phase diagram data in the BaO-MoO_3 pseudo-binary section

There is no source in the literature reporting (to the best of our knowledge) a phase diagram over the complete composition range between BaO and MoO_3. BaMoO_4 is found on the BaO-MoO_3 pseudo-binary section at the composition \( x(\text{MoO}_3) = 0.5 \). Phase equilibrium data in the BaO-BaMoO_4 section were reported by Yanushkevich and Zhukovskii [68], while a sketch of the BaMoO_4-MoO_3 section was reported by Ustinov et al. [56] and Zhukovskii et al. [69]. However, the latter two sets of data are in rather poor agreement as seen in Fig. 12.

Yanushkevich and Zhukovskii studied the BaO-BaMoO_4 section using X-ray and visual polythermal techniques [68] on BaCO_3-MoO_3 mixtures after thermal treatment. Two intermediate compounds were identified on the pseudo-binary section, namely Ba_2MoO_5 and Ba_3MoO_6, with transition temperatures \( T = 1573 \text{ K} \) (peritectic) and \( T \sim 1825 \text{ K} \) (congruent melting), respectively. In addition the authors found two eutectic equilibria, i.e. between BaO and Ba_3MoO_6 around 1808 K, and between Ba_2MoO_5 and Ba_3MoO_6 around 1493 K.

Ustinov et al. investigated the BaMoO_4-MoO_3 phase diagram using thermography and X-ray diffraction [56] on (BaO:MoO_3) mixtures. The authors reported the existence of an incongruent melting compound of BaMoO_3 (with a decomposition temperature around 988 K), and an eutectic equilibrium at \( x(\text{MoO}_3) = 0.811 \) and \( T = 913 \text{ K} \). The interplanar spacings reported for the BaMoO_3 phase seem to match the BaMoO_3 phase data collected in this work for BaMoO_3-rich-side were found much higher than in the studies of Ustinov et al., and seem more trustworthy based on the expected melting temperature of the barium molybdate.

2.5. Thermodynamic modelling assessment

Dash et al. [53] reported a thermodynamic model for the Ba-Mo-O system using the SOLGASMIX-PV program [70]. The authors considered 9 compounds in this system: BaMoO_3, BaMoO_4, BaMoO_5, BaMoO_6, BaMoO_7, BaMoO_8, BaMoO_9, and BaMoO_10. The Gibbs energy functions for the ternary phases were estimated using the additive oxide method for the standard entropy and enthalphy of formation, and using the rule of Neumann-Kopp for the heat capacity, when no experimental data were available in the literature. Only the computed isothermal section of the Ba-Mo-O phase diagram at \( T = 700 \text{ K} \) is shown in their work. The authors reported \( O(\text{Ba}), Ba(\text{g}), \text{BaMoO}_4(\text{g}) \) and \( \text{BaMoO}_3(\text{g}) \) to be the predominant vapour species in all ternary phase fields, and over pure compounds. The authors also observed that the calculated oxygen potentials over the ternary phase fields and pure compounds were not affected much by the change of the Gibbs energy functions of the ternary phases. However, this had a more pronounced effect on the partial pressures of Ba(g), BaO(g), BaMoO_4(g) and BaMoO_3(g) (one to two orders of magnitude). Experimental investigations of the stable ternary phase fields are needed to confirm the predictions of the computed phase diagram by [53], which are unfortunately not available to this date.

3. Experimental methods

3.1. Sample preparation

BaMoO_4 was prepared by solid state reaction between accurately weighted quantities of barium carbonate (BaCO_3, >99%, Fluka) and molybdenum oxide (MoO_3, 99.95%, Sigma–Aldrich), heated under air in an alumina boat at 1023 K for 24 h with an intermediate regrinding step. BaMoO_3 was subsequently obtained by reduction of the synthesized BaMoO_4 material in Ar/H_2(5\%) flow at 1130 K for 12 h in a tubular furnace. BaMoO_4 was synthesized by mixing BaMoO_3 and MoO_3 in stoichiometric ratio and heating in a platinum crucible under oxygen flow for 12 h at 823 K, followed by 2 h at 993 K. After regrinding, the sample was subjected to a second thermal treatment for 12 h at 823 K.

The purity of the synthesized materials was checked using X-ray diffraction. No secondary phases were detected, and the samples'
provinces is expected to be better than 99% 4 .

3.2. Powder X-ray diffraction (XRD)

X-ray diffraction patterns were collected at room temperature using a PANalytical X’Pert PRO X-ray diffractometer mounted in the Bragg-Brentano configuration with a Cu anode (0.4 mm × 12 mm line focus, 45 kV, 40 mA), and the X-ray scattered intensities were measured with a real time multi strip (RTMS) detector (X’Celerator). The data were collected by step scanning in the angle range 10 ≤ θ ≤ 120° with a step size of 0.008° (2θ); total measuring time was about 8 h. Structural analysis was performed by the Rietveld method with the FullProf suite [71].

3.3. X-ray Absorption Near Edge Structure Spectroscopy (XANES)

XANES measurements were performed at the BM26A-DUBBLE BeamLine (Dutch-Belgian Beamline) of the European Synchrotron Radiation Facility (ESRF, Grenoble, France). Small amounts (10–20 mg) of powdered sample were mixed with boron nitride (BN), and pressed into pellets for the measurements. The storage ring operating conditions were 6.0 GeV and 170–200 mA. A double crystal monochromator mounted with a Si(111) crystal coupled to collimating and focusing Pt coated mirrors was used.

XANES spectra were collected at room temperature in transmission mode at the Mo K-edge. A step size of 1 eV was used in the edge region. The energy E0 of the edge absorption threshold position was taken at the inflection point of the spectrum by using the first node of the second derivative. The position of the pre-peak was selected from the first node of the first derivative. Several acquisitions were performed on the same sample and summed up to improve the signal-to-noise ratio. Before averaging the scans, each spectrum was aligned using the XANES spectrum of a metallic molybdenum reference foil measured before and after the series of samples under investigation (i.e. MoO2, MoO3, BaMoO4, and BaMoO5). The ATHENA software [72] was used to normalize the spectra.

3.4. Low-temperature heat capacity

Low-temperature heat capacity measurements were performed on BaMoO4 (m = 12.25(5) mg) and BaMoO3 (m = 40.63(5) mg) in the temperature range T = (2.1–293.2) K (BaMoO4), and T = (3.0–293.2) K (BaMoO3), respectively, using a PPMS (Physical Property Measurement System, Quantum Design) instrument, with no applied magnetic field (B = 0 T). The samples were encapsulated in Stycast 2850 FT for the measurements (mSt = 2.87 mg and mSt = 3.79 mg for BaMoO4 and BaMoO3, respectively), so as to improve the heat transfer with the sample platform of these oxide materials. The heat capacity contribution of the Stycast was subtracted from the recorded data. This technique is based on a relaxation method, which was critically assessed by Lashley et al. [73]. The contributions of the sample platform, wires, and grease were deduced by a separate measurement of an addenda curve. Based on the experience acquired on this instrument with standard materials and other compounds, and the error associated with the encapsulation procedure in Stycast [74], the uncertainty was estimated at about 1–2% in the middle range of acquisition (from 10 to 70 K), and reaching about 3% at the lowest temperatures and near room temperature [73,74].

3.5. Differential Scanning Calorimetry (DSC)

The transition temperatures in the BaMoO4-MoO3 pseudo-binary section of the Ba-Mo-O system were measured using simultaneous Thermogravimetry (TG)-Differential Scanning Calorimetry (DSC) measurements using plate type sensors in the TG-DSC module of the Setaram 96 Line calorimeter.

The temperatures were monitored throughout the experiments by a series of interconnected S-type thermocouples. The temperature on the heating ramp (10 K min−1) was calibrated and corrected for the effect of the heating rate by measuring the melting points of standard high purity metals (In, Sn, Pb, Al, Ag, Au) at 2-4-6-8-10-12 K min−1. The calibration procedure was performed as recommended by Hène et al. [75] and Gatta et al. [76]. The transition temperatures in the BaMoO4-MoO3 phase diagram were derived on the heating ramp as the onset temperature using tangential analysis of the recorded heat flow. The liquidus temperature of mixtures was derived from the peak extremum of the last thermal event. The uncertainty on the measured temperatures is estimated to be ± 5 K for pure compounds and ±10 K for mixtures.

BaMoO4 and MoO3 were mixed together in the selected stoichiometric ratios by grinding at room temperature, and subsequently inserted in the calorimeter for measurement. In some cases, the mixtures were pre-treated under air at 800 K for 20 h before the TG-DSC measurement. The samples were placed in an alumina crucible on top of

Table 4

Provenance and purity of the samples investigated in this study. XRD: X-ray diffraction; DSC: Differential Scanning Calorimetry.

| Formula | Source          | State      | Color           | Mass fraction purity | Purity analysis method |
|---------|----------------|------------|-----------------|----------------------|------------------------|
| MoO3    | Sigma-Aldrich  | Powder     | Light yellow/green | 0.9995 ± 0.0005    | Provided by supplier   |
| BaMoO3  | Fluka          | Powder     | White           | >0.99 ± 0.01        | Provided by supplier   |
| BaMoO4  | Synthesized    | Powder     | White           | 0.99 ± 0.01         | XRD                    |
| BaMoO3  | Synthesized    | Powder     | Red             | 0.99 ± 0.01         | XRD                    |
| BaMoO5  | Synthesized    | Powder     | White           | 0.99 ± 0.01         | XRD, DSC               |

Table 5

Refined lattice parameters. X-ray diffraction measurements were performed at room temperature and atmospheric pressure. The derived standard uncertainties are given in parenthesis.

| Phase      | Symmetry  | Space group | a/\(\text{nm}\) | b/\(\text{nm}\) | c/\(\text{nm}\) | \(\beta/\text{o}\) | \(\rho/\text{g \cdot cm}^{-3}\) |
|------------|-----------|-------------|----------------|----------------|----------------|----------------|-----------------------------|
| BaMoO4     | Tetragonal| I4_1/a      | 0.5583(1)      | 0.5583(1)      | 1.2822(4)      | 90             | 4.9399(2)                  |
| BaMoO3     | Cubic     | Pn\text{3m} | 0.4043(1)      | 0.4043(1)      | 0.4041(3)      | 90             | 7.0771(3)                  |
| BaMoO5\(_{10}\) | Monoclinic | P2\(_1\)    | 1.4696(3)      | 0.7568(1)      | 0.6965(1)      | 100.43(1)      | 5.101(1)                   |

Table 6

Energies of the pre-peaks and inflection points of the Mo K-edge XANES spectra.

| Compound | Pre-peak (eV) | Inflection point E\(_\text{I}\) (eV) |
|----------|--------------|---------------------------------|
| Mo\(_0\) | 2000.0       |                                 |
| Mo\(_0\)O\(_2\) | 2001.2       |                                 |
| Mo\(_0\)O\(_3\) | 2000.7       |                                 |
| BaMoO3   | 2000.9       |                                 |
| BaMoO4   | 2000.1       |                                 |
| BaMoO5   | 2000.6       |                                 |

\(a\) The estimated expanded uncertainty (with a coverage factor \(k = 2\)) on the energies is 1.0 eV.
boron nitride powder to avoid chemical interactions with the crucible upon melting (which would lead to the formation of aluminium molybdate). They were measured under oxygen flow to avoid reduction of the MoO$_3$ to lower valence states molybdenum oxides. It should be noted that with the present measurement configuration, the data could be collected up to a maximum temperature of about 1173 K, due to excessive boron nitride oxidation above the latter temperature, which affected the shape of the heat flow baseline curve. One typical measurement consisted in three heating cycles with 10 K \cdot min$^{-1}$ heating rate. The data collected on the first heating ramp were not considered for the analysis, however. The first heating cycle was used to equilibrate the samples, and the data collected in the subsequent two cycles were used for the analysis. The shape of the heat flow signal was mostly identical on the second and third cycles, indicating that thermodynamic equilibrium conditions were reached. After the TG-DSC measurements, selected samples were analysed using X-ray diffraction to confirm the nature of the observed transitions.

In addition, the enthalpy associated with the peritectic decomposition of BaMo$_2$O$_{10}$ was determined in this work using the Multi HTC module of the same 96 Line allowing 3D-heat flow measurements. The enthalpy was determined by placing a reference material of well-known transition enthalpy in the reference crucible and measuring both sample and reference materials in the same cycle. This configuration allows to calculate for each individual measurement cycle the detector sensitivity equal to:

$$s_{\text{ref}} = \frac{M_{\text{ref}} \cdot A_{\text{ref}}}{m_{\text{ref}} \cdot \Delta_H(T_{\text{ref}})}$$

where $s_{\text{ref}}$ is the detector sensitivity in $\mu V \cdot mW^{-1}$, $M_{\text{ref}}$ the molar mass in g \cdot mol$^{-1}$, $m_{\text{ref}}$ the weight of the reference in mg, $A_{\text{ref}}$ the peak area corresponding to the transition event in $\mu V \cdot s$, and $\Delta_H(T_{\text{ref}})$ the enthalpy of transition of the reference material in J \cdot mol$^{-1}$.

The detector sensitivity is assumed to remain the same at the temperature of the transition event of the sample, which is a reasonable approximation for two events sufficiently close to each other.

4. Thermodynamic models

The thermodynamic parameters of the Ba-Mo-O CALPHAD model were optimized using the PARROT module of the Thermo-Calc software (Version 2016b) [77,78]. The hexavalent barium molybdates BaMoO$_4$, BaMoO$_3$, Ba$_2$MoO$_5$, BaMo$_2$O$_7$, and BaMo$_3$O$_{10}$ were treated as stoichiometric compounds. BaMoO$_3$ was described with a more general formalism for a perovskite phase, compatible with the description of the grey-phase (Ba,Sr)(Zr,U,Mo)O$_3$ in the TAF-ID database [10]. The Gibbs energy functions of all the phases are referred to the enthalpy of the pure elements in their stable state at room temperature 298.15 K and 1 bar ("H$^{\text{SER}}_{298.15K}$"). Note that the gas phase included in the present model was not optimized, and will be the subject of future works. The optimized parameters of the binary constituting sub-systems Ba-Mo and Ba-O can be found in Tables B.1, B.2, C.1, and C.2.
4.1. Pure elements

The Gibbs energy functions of the pure elements \( i \) at temperature \( T \) and in their state \( \varphi \) are given by:

\[
G_i^\varphi(T) = -n_i^\varphi H_i^\varphi(298.15 \, K) + a + bT + cT \ln T + \sum d_i T^n
\]

where \( n \) is an integer (2, 3, -1...). The parameters reported by Dinsdale are used in this work for pure barium, molybdenum, and oxygen [79].

Metallic barium and molybdenum are included in the description of the BCC A2 phase with sublattices \((\text{Ba}_2\text{Mo}_4)(\text{O}_3\text{Va}_3)\) (\( \text{Va} \) being a vacancy). The Gibbs energy functions of the \((\text{Ba}_2\text{Va})_3\) and \((\text{Mo}_3\text{Va})_3\) end-members are those of Dinsdale [79].

4.2. Binary oxides

The binary oxides \( \text{BaO}_2, \text{MoO}_2, \text{Mo}_2\text{O}_{11}, \text{Mo}_3\text{O}_{23}, \text{Mo}_9\text{O}_{26}, \text{and MoO}_3 \) are described as stoichiometric compounds. The corresponding Gibbs energy functions have the same form as in Eq. (7):

\[
G_i^\varphi(T) = -n_i^\varphi H_i^\varphi(298.15 \, K) + a + bT + cT \ln T + \sum d_i T^n
\]

where \( n_i^\varphi \) is the number of atoms of the \( i \)th element in the oxide formula. These are taken from the TAF-ID database [10].

4.3. Halite phase

The halite \( \text{BaO} \) is described with a two sublattice model \((\text{Ba}^{2+}, \text{Va})(\text{O}^{2-}, \text{Va})\), where \( \text{Va} \) are vacancies. The Gibbs energies of the end-members take the same form as in Eq. (7) and Eq. (8). No interaction parameter was introduced.

4.4. Hexavalent ternary molybdates

The hexavalent ternary molybdates \( \text{Ba}_2\text{MoO}_6, \text{Ba}_3\text{MoO}_9, \text{Ba}_2\text{MoO}_5, \text{Ba}_2\text{Mo}_2\text{O}_7, \text{and BaMo}_3\text{O}_{10} \), of general formula \( \text{Ba}_m\text{Mo}_n \), are described with a three-sublattice model of the form \((\text{Ba}^{2+})_m(\text{Mo}^{6+})_n(\text{O}^{2-})_3\). The Gibbs energy functions of the corresponding phases have the same form as in Eq. (7) and Eq. (8).

Experimental thermodynamic data are only available for the \( \text{BaMoO}_3 \) scheelite phase as detailed in Section 2. The Gibbs energy for this phase has been expressed based on the recommended enthalpy of formation, standard entropy and heat capacity, and the enthalpic \( a \) and entropic \( b \) coefficients were subsequently optimized to fit the reported congruent melting temperature and the rest of the thermodynamic and phase diagram data in the \( \text{BaO-MoO}_3 \) section. For the other compounds, the Gibbs energies have been expressed as a function of \( \text{BaMoO}_4 \) and the binary oxides \( \text{BaO} \) and \( \text{MoO}_3 \). The corresponding enthalpies of formation and entropies have been further optimized to fit the available phase diagram data in the \( \text{BaO-MoO}_3 \) pseudo-binary section.

4.5. \( \text{BaMoO}_3 \) perovskite

The descriptions adopted for the \( \text{BaUO}_3 \) and \( \text{BaZrO}_3 \) perovskites in the TAF-ID database is also adopted in this work for \( \text{BaMoO}_3 \) in order to extrapolate to higher order systems, and the modelling of the multi-component grey-phase formed in irradiated fuel \( \text{Ba(Zr,U,Mo)O}_3 \). A three-sublattice model is used with \( \text{Ba}^{2+} \) and \( \text{Mo}^{4+} \) cations on the second sublattice, i.e. \((\text{Ba}^{2+})_n(\text{Mo}^{4+})_n(\text{O}^{2-})_3\). The corresponding higher-order model for the grey-phase in the TAF-ID is \((\text{Ba}^{2+})_n(\text{Zr}^{4+})_n(\text{U}^{4+})_n(\text{O}^{2-})_3\). The partial substitution of the \( \text{U}^{4+} \) by \( \text{Ba}^{2+} \) on the second sublattice allows to reproduce the stability of the perovskite along the \( \text{BaUO}_3-\text{BaZrO}_3 \), and \( \text{BaMoO}_3 \) tie-line in the \( \text{Ba-U-O} \) system [10]. The Gibbs energy functions of the \((\text{Ba}^{2+})_n(\text{Mo}^{4+})_n(\text{O}^{2-})_3 \) and \((\text{Ba}^{2+})_n(\text{Ba}^{2+})_n(\text{O}^{2-})_3 \) end-members take the same forms as in Eq. (7) and Eq. (8).

4.6. Liquid phase

An ionic two-sublattice model is used to describe the liquid phase [80], with \( \text{Ba}^{2+} \), and \( \text{Mo}^{4+} \) cations on the first sublattice, and \( \text{MoO}_2^2- \), \( \text{O}^{2-} \) anions, charged vacancies \( \text{Va}^{2+} \), neutral \( \text{MoO}_3 \), and neutral oxygen \( \text{O} \) on the second sublattice:

\[
(\text{Ba}^{2+}, \text{Mo}^{4+})_m(\text{MoO}_2^2-, \text{O}^{2-}, \text{Va}^{2+}, \text{MoO}_3, \text{O})_n\]

where \( y\text{Ba}^{2+}, y\text{Mo}^{4+}, y\text{MoO}_2^2-, y\text{O}^{2-}, \) and \( y\text{Va}^{2+} \) are the site fractions of barium cations, molybdenum cations, \( \text{MoO}_2^2- \), oxygen anions, and charged vacancies on the second sublattice, respectively. \( P \) and \( Q \) vary with composition via the site fractions so as to keep the phase electrically neutral.

The Gibbs energy of the liquid phase in this formalism is given by:

\[
G_{\text{liquid}} = y\text{Ba}^{2+}y\text{Mo}^{4+}G(\text{Ba}^{2+})_m(\text{Mo}^{4+})_n + y\text{Ba}^{2+}y\text{O}^{2-}G(\text{Ba}^{2+})_m(\text{O}^{2-})_n + y\text{Mo}^{4+}y\text{MoO}_2^2-G(\text{Mo}^{4+})_m(\text{MoO}_2^2)_n + y\text{O}^{2-}y\text{O}-G(\text{O}^{2-})_m(\text{O})_n
\]

\[
G_{\text{liquid}} = y\text{Ba}^{2+}y\text{Mo}^{4+}G(\text{Ba}^{2+})_m(\text{Mo}^{4+})_n + y\text{Ba}^{2+}y\text{O}^{2-}G(\text{Ba}^{2+})_m(\text{O}^{2-})_n + y\text{Mo}^{4+}y\text{MoO}_2^2-G(\text{Mo}^{4+})_m(\text{MoO}_2^2)_n + y\text{O}^{2-}y\text{O}-G(\text{O}^{2-})_m(\text{O})_n
\]

\[
G_{\text{liquid}} = y\text{Ba}^{2+}y\text{Mo}^{4+}G(\text{Ba}^{2+})_m(\text{Mo}^{4+})_n + y\text{Ba}^{2+}y\text{O}^{2-}G(\text{Ba}^{2+})_m(\text{O}^{2-})_n + y\text{Mo}^{4+}y\text{MoO}_2^2-G(\text{Mo}^{4+})_m(\text{MoO}_2^2)_n + y\text{O}^{2-}y\text{O}-G(\text{O}^{2-})_m(\text{O})_n
\]

\[
G_{\text{liquid}} = y\text{Ba}^{2+}y\text{Mo}^{4+}G(\text{Ba}^{2+})_m(\text{Mo}^{4+})_n + y\text{Ba}^{2+}y\text{O}^{2-}G(\text{Ba}^{2+})_m(\text{O}^{2-})_n + y\text{Mo}^{4+}y\text{MoO}_2^2-G(\text{Mo}^{4+})_m(\text{MoO}_2^2)_n + y\text{O}^{2-}y\text{O}-G(\text{O}^{2-})_m(\text{O})_n
\]

\[
G_{\text{liquid}} = y\text{Ba}^{2+}y\text{Mo}^{4+}G(\text{Ba}^{2+})_m(\text{Mo}^{4+})_n + y\text{Ba}^{2+}y\text{O}^{2-}G(\text{Ba}^{2+})_m(\text{O}^{2-})_n + y\text{Mo}^{4+}y\text{MoO}_2^2-G(\text{Mo}^{4+})_m(\text{MoO}_2^2)_n + y\text{O}^{2-}y\text{O}-G(\text{O}^{2-})_m(\text{O})_n
\]
Fig. 6. Normalized XANES spectra of BaMoO₄ and BaMoO₃ together with Mo, MoO₂ and α-MoO₃ reference materials.

4.7. Gas phase

The gas phase is described by an ideal mixture of (Ba, Ba₂, BaO, Ba₂O₃, Ba₂O₅, Mo, Mo₂, MoO, MoO₂, Mo₂O₆, Mo₃O₈, Mo₄O₁₂, Mo₅O₁₅, BaMoO₄, O, O₂, O₃) gaseous species. The Gibbs energy is expressed by:

\[ G^o = \sum_i \gamma_i G^o_i + RT \sum_i \gamma_i \ln \gamma_i + RTh_F / P^o \]  

where \( \gamma_i \) is the fraction of the species \( i \) in the gas phase, \( G^o_i \) the standard Gibbs energy of the gaseous species \( i \), and \( P^o \) the standard pressure. The function for BaMoO₄ is taken from the SGTE database [81]. Note that the computed pressures in the ternary system Ba-Mo-O are not commented in this work, and will be the subject of future work in our research group. We refer the reader to the Appendix for the expressions of the gaseous functions of the binary species.

5. Experimental results and discussion

5.1. Structural analysis

BaMoO₄, BaMoO₃, and BaMo₂O₁₀ adopt a tetragonal, cubic, and monoclinic structure, respectively. The scheelite BaMoO₄ crystallizes in space group \( \text{I} \overline{4} \text{a} \). BaMoO₃ shows a perovskite-type structure, in space group \( \text{Pn} \overline{3} \text{m} \). BaMo₂O₁₀ was reported with the space group \( \text{P}2_1 \). The refined cell parameters obtained by the Rietveld method from the XRD data are summarized in Table 5, and the X-ray diffraction patterns are shown in Fig. 5. The refined atomic positions are provided in the Supplementary Information.

5.2. Valence state determination by XANES: BaMoO₄ and BaMoO₃

The XANES spectra of BaMoO₄ and BaMoO₃ collected at the Mo K-edge are shown in Fig. 6 together with Mo⁰, Mo⁴O₂, and α-Mo⁴O₃ reference materials. The inflection point \( E_0 \) positions (corresponding to the absorption edges) are listed in Table 6. The absorption edge of BaMoO₄ is very close to that of Mo⁴O₂, while that of BaMoO₃ is very well aligned with that of Mo⁴O₃, confirming the tetravalent and hexavalent valence states of molybdenum in those two materials, respectively. A shift of the inflection point to higher energy is observed with increasing valence state, as expected. In addition, the spectrum of BaMoO₄ shows a characteristic pre-edge feature around 2000 eV of relatively high intensity. This is due to the presence of short and highly covalent Mo-O bonds in tetrahedral geometry in the latter compound (see Fig. 5a), which enhance 4d-5p mixing through their hybridization with O(2p) [82–84]. Similarly, a pre-edge shoulder appears in the spectrum of α-MoO₃ related to the distorted MoO₄ octahedra in the structure (the \( 1\alpha \text{Mo} \rightarrow 4\text{d(Mo)} + 2\text{p(O)} \) transition is dipole-forbidden.
measured in the absence of magnetic field are shown in Figs. 2 b and 8 a, for a perfectly regular MoO$_4$ octahedron. A similar pre-peak feature has been reported in the literature for Na$_4$MoO$_4$, K$_2$MoO$_4$, and CaMoO$_4$ [84].

5.3. Standard entropy determination: BaMoO$_4$ and BaMoO$_3$

The low-temperature heat capacity data of BaMoO$_4$ and BaMoO$_3$ measured in the absence of magnetic field are shown in Figs. 2 b and 8 a, respectively, and are listed in the Supplementary Information. The heat capacity of both materials increases smoothly with temperature. The collected data do not exhibit any noticeable anomaly. The results for BaMoO$_3$ are slightly higher than the data collected by Morishita et al. in the temperature range (2.02–297.19) K with the same relaxation method. Two measurements were performed in this work for BaMoO$_4$ with two different samples (12.25 mg and 22.85 mg, respectively). The data obtained with the 22.85 mg pellet were found in very good agreement with Morishita et al., but the other data were preferred for the reasons detailed hereafter. The data for BaMoO$_3$ are reported for the first time in this work.

The thermodynamic functions of BaMoO$_4$ and BaMoO$_3$ were derived at 298.15 K by fitting the experimental data using the OriginPro 2015 software with the Levenbergh Marquardt iteration algorithm, using Origin C type fitting functions in the high temperature region, and simple expression type functions in the low-temperature region. The fitted data are shown with solid lines in Figs. 7a, 8a and 7b, 8b.

In the low-temperature region (below $T = 15.7 \text{ K}$ for BaMoO$_4$ and $T = 26.6 \text{ K}$ for BaMoO$_3$ in this case), the phonon contribution can be modelled using an harmonic-lattice model [85], as given by Eq. (14), where the number of required terms augments with the high-temperature limit of the fit:

$$C_{\text{lat}} = \sum B_n T^n, \quad \text{where} \quad n = 3, 5, 7, 9...$$

(14)

The corresponding coefficients for BaMoO$_4$ and BaMoO$_3$ are listed in Table 7.

The electronic contribution of the conduction electrons at the Fermi surface are expressed with a linear term $\gamma T$ [86]. The electronic specific heat of BaMoO$_3$ is zero, as could be expected for such an insulating material. An electronic contribution of 7.71 mJ \cdot mol$^{-1} \cdot$K$^{-2}$ is found for BaMoO$_3$. This is in good accordance with the study of Hayashi and Aoki [87], who reported metallic conductivity in the range (2.5 to 300) K based on resistivity measurements.

In the high-temperature region, the lattice contribution is modelled using a combination of Debye and Einstein functions [88], as expressed in Eq. (15). This method was used in the literature for several classes of inorganic materials: iron phosphates [89–91], zirconolite [92], calcium titanate [93], dicesium molybdate [94], double molybdates [95,96], alkali uranate and neptunate [97,98]. Two Einstein functions were used in combination with a Debye function. The fitted parameters are listed in Table 7. The sum ($n_0 + n_{E1} + n_{E2}$) is 5.97 and 5.08 for BaMoO$_4$ and BaMoO$_3$, respectively. The same procedure applied to the 22.85 mg pellet of BaMoO$_4$ gave a sum ($n_0 + n_{E1} + n_{E2}$) equal to 5.35, hence an underestimation compared to the 6 atoms expected from the formula unit. The measurement of very insulating materials as is the case for BaMoO$_4$ is not always straightforward using the thermal-relaxation technique. The encapsulation procedure in Stycast has proved

### Table 7

| Material      | Harmonic lattice-model | Debye and Einstein fit |
|---------------|------------------------|------------------------|
|               | Temp. range/K          | Temp. range/K          |
| BaMoO$_4$     | 2.1–18.7               | 16.9–293.2             |
|               | $\gamma$/mJ mol$^{-1}$ K$^{-2}$ | $n_0$/mol     |
|               | $B_0$/mJ mol$^{-1}$ K$^{-4}$ | $\theta_d$/K        |
|               | $B_0$/mJ mol$^{-1}$ K$^{-6}$ | $n_{E1}$/mol     |
|               | $B_0$/mJ mol$^{-1}$ K$^{-8}$ | $\theta_{E1}$/K      |
|               | $B_0$/mJ mol$^{-1}$ K$^{-10}$ | $n_{E2}$/mol      |
|               | $B_0$/mJ mol$^{-1}$ K$^{-12}$ | $\theta_{E2}$/K     |
| BaMoO$_3$     | 1.9–26.6               | 24.9–293.2             |

Fig. 8. Heat capacity of BaMoO$_4$ (○) measured in zero magnetic field and fit to the data (red line). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)
effective to improve the thermal coupling with the sample platform. Nevertheless, a too large weight in case of an insulator can lead to strong thermal inertia and long relaxation times, which reduces the accuracy of the results above \( \sim 250 \) K. Because the fitting results obtained on the 12.25 mg pellet were found closer to the expected 6 atoms per formula unit than the results obtained on the 22.85 mg pellet, the former data were considered to be more reliable and were selected in this work. As a consequence, the retained heat capacity and entropy data for the thermodynamic assessment are higher than reported in the work of Morishita et al. \(^{49}\). It should be pointed out that the optimized standard entropy for BaMoO\(_4\) in the thermodynamic model (see Section 6) is in very good agreement with the data selected herein. In fact, optimization of the thermodynamic model while constraining the heat capacity and standard entropy to values closer to that of Morishita et al. was also tested, but did not give as satisfactory results.

\[
C_{p,n} = n_D D(\theta_0) + n_{E1} E(\theta_{E1}) + n_{E2} E(\theta_{E2})
\]

(15)

where \( D(\theta_0) \), \( E(\theta_{E1}) \), and \( E(\theta_{E2}) \) are the Debye and Einstein functions, respectively, as written in equations (16) and (17). \( \theta_0 \), \( \theta_{E1} \), and \( \theta_{E2} \) are the characteristic Debye and Einstein temperatures. \( n_D \), \( n_{E1} \), and \( n_{E2} \) are adjustable parameters, whose sum (\( n_D + n_{E1} + n_{E2} \)) should be approximately equal to the number of atoms in the formula unit (i.e., 6 and 5 in this case).

\[
D(\theta_0) = \frac{9 \hbar^2}{(\theta_0^2)^3} \int_0^{\theta_0/\theta} \frac{e^x x^4}{[e^x - 1]^4} \, dx
\]

(16)

\[
E(\theta_E) = 3 \pi R^2 \frac{e^{\theta_0/\theta}}{[e^{\theta_0/\theta} - 1]^2}, \quad x = \frac{\theta_E}{\theta}
\]

(17)

where \( R \) is the universal gas constant.

The heat capacity values at 298.15 K obtained by extrapolation are \( C_{p,n}^{\text{eq}}(\text{BaMoO}_4, \text{cr}, 298.15 \, \text{K}) = (126.4 \pm 3.8)^2 \, \text{J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \) and \( C_{p,n}^{\text{eq}}(\text{BaMoO}_3, \text{cr}, 298.15 \, \text{K}) = (103.4 \pm 3.1)^3 \, \text{J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \). The experimental standard entropies at 298.15 K determined by numerical integration of \( (C_{p,n}/T)f(T) \) using the aforementioned fitted functions, are \( S_{n}^{\text{eq}}(\text{BaMoO}_4, \text{cr}, 298.15 \, \text{K}) = (160.4 \pm 3.8)^4 \, \text{J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \) and \( S_{n}^{\text{eq}}(\text{BaMoO}_3, \text{cr}, 298.15 \, \text{K}) = (111.3 \pm 2.8)^5 \, \text{J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \).

5.4. Phase diagram measurements in the BaMoO\(_4\)-MoO\(_3\) pseudo-binary section

The transition temperatures in the BaMoO\(_4\)-MoO\(_3\) pseudo-binary system measured in this work by TG-DSC and DSC are listed in Table 8, and shown in Figs. 12 and 13. The corresponding thermograms are shown in Figs. A.1 and A.2 in the Appendix. No noticeable weight loss was observed from the thermogravimetry results, thus the initial

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**Table 8**

| (BaMoO\(_3\)) | (MoO\(_3\)) | \( T_m/K \) | Equilibrium | Equilibrium reaction | Post-XRD |
|---------------|-------------|-------------|-------------|----------------------|---------|
| (0.59;0.41)   | 0.629       | 894 \pm 10^2| Peritectoid  | BaMoO\(_3\) + BaMoO\(_4\) + BaMoO\(_3\)| -       |
|               | –           | 915 \pm 10^2| Peritectic   | BaMoO\(_3\) + BaMoO\(_4\) | –       |
| (0.53;0.461)  | 0.65        | 894 \pm 10^2| Peritectoid  | BaMoO\(_3\) + BaMoO\(_4\) + BaMoO\(_3\)| –       |
|               | –           | 917 \pm 10^2| Peritectic   | BaMoO\(_3\) + BaMoO\(_4\) | –       |
| (0.5;0.5)     | 0.6667      | 887 \pm 10^2| Peritectic   | BaMoO\(_3\) + BaMoO\(_4\) + BaMoO\(_3\)| –       |
| (0.28;0.72)   | 0.781       | 897 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |
| (0.25;0.75)   | 0.8         | 888 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |
| (0.21;0.79)   | 0.826       | 897 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |
| (0.176;0.824) | 0.85        | 897 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |
| (0.15;0.85)   | 0.87        | 895 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |
| (0.11;0.89)   | 0.9         | 886 \pm 10^2| Eutectic     | BaMoO\(_3\) + Liq. – MoO\(_3\) | BaMoO\(_3\) + MoO\(_3\) |

\(^{a}\) Data measured by TG-DSC.

\(^{b}\) Data measured by DSC.

\(^{c}\) Onset temperature of the heat flow signal.

\(^{d}\) Peak maximum of the heat flow signal.

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\(^{2}\) The quoted uncertainty corresponds to the standard uncertainty.

\(^{3}\) The quoted uncertainty corresponds to the standard uncertainty.
The measured temperatures were interpreted with an eutectic equilibrium at $T = 983$ K and $x(\text{MoO}_2) = 0.83$. A single event was observed for the samples of compositions $x(\text{MoO}_2) = 0.826$ and 0.833. Moreover, BaMoO$_{10}$ ($x(\text{MoO}_2) = 0.75$) is suggested to decompose in a peritectic reaction. The measured equilibria points around $T = 892$ K are attributed to the peritectoid decomposition of BaMo$_2$O$_7$. The latter interpretation is based on a limited number of measured points. To confirm this interpretation, complementary measurements coupled with post-XRD and microscopic characterizations would be very valuable. It is worth pointing out that the reported decomposition temperature of BaMo$_2$O$_7$ by Zhukovskii et al. [69] matches well the measured peritectic decomposition of BaMo$_2$O$_{10}$.  

5.5. Transition enthalpy determination: BaMo$_2$O$_{10}$

The decomposition temperature and enthalpy associated with the peritectic decomposition of BaMo$_2$O$_{10}$ was determined using Differential Scanning Calorimetry. The average transition temperature over four heating cycles yielded (918 ± 5) K. In addition, the enthalpy of transition was estimated by measuring BaMo$_2$O$_{10}$ together with Cs$_2$TeO$_4$.
Fig. 10. (a) Enthalpy increments of BaMoO$_3$ measured experimentally and calculated in the model. (b) Heat capacity of BaMoO$_3$ measured experimentally and calculated in the model.

Fig. 11. (a) Enthalpy increments of BaMoO$_4$ measured experimentally and calculated in the model. (b) Heat capacity of BaMoO$_4$ measured experimentally and calculated in the model.

Fig. 12. BaO-MoO$_3$ pseudo-binary phase diagram computed from the thermodynamic model, and comparison with the experimental data.

Fig. 13. BaMoO$_4$-MoO$_3$ pseudo-binary phase diagram computed from the thermodynamic model, and comparison with the experimental data.
reference material, whose enthalpy of transition was determined in another work in our research group as \( \Delta_{\text{tr}} H_{\text{om}}(\text{Cs}_2\text{TeO}_4, \text{cr}, T_{\text{tr}}) = (2.67 \pm 0.14) \text{ kJ/mol} \) \cite{99}. \text{Cs}_2\text{TeO}_4 was chosen as it displays a transition temperature close to the peritectic transition of \text{BaMo}_3\text{O}_{10}, but without overlapping. Moreover, both reference and sample being oxide materials, they are expected to show similar thermal properties (heat capacity and thermal conductivity). The curve of the recorded heat flow versus temperature for the latter measurement is shown in Fig. 9.

The first single peak corresponds to the polymorphic transition of \text{Cs}_2\text{TeO}_4 (from an orthorhombic \( \alpha \) phase to an hexagonal \( \beta \) phase \cite{99}) and the second peak to the peritectic decomposition of \text{BaMo}_3\text{O}_{10}. Note that the opposite directions for both (endothermic) events is due to the different positioning in the reference and sample crucibles, respectively.

| Invariant equilibrium | Equilibrium type | CALPHAD model | Exp. data |
|-----------------------|-----------------|---------------|-----------|
| \text{BaO} + \text{Ba}_3\text{MoO}_6 \rightarrow \text{Liquid} | Eutectic | \( x(\text{MoO}_3) = 0.19 \) | \( T/K = 1794 \) | \( x(\text{MoO}_3) = 0.175 \) |
| \text{Ba}_3\text{MoO}_6 \rightarrow \text{Liquid} | Congruent melting | \( x(\text{MoO}_3) = 0.25 \) | \( T/K = 1833 \) | \( x(\text{MoO}_3) = 0.25 \) |
| \text{Ba}_3\text{MoO}_6 + \text{BaMoO}_4 \rightarrow \text{Liquid} | Peritectic | \( x(\text{MoO}_3) = 0.3333 \) | \( T/K = 1572 \) | \( x(\text{MoO}_3) = 0.3333 \) |
| \text{BaMo}_2\text{O}_7 \rightarrow \text{BaMoO}_4 + \text{Ba}_{2}\text{MoO}_7 | Peritectoid | \( x(\text{MoO}_3) = 0.6667 \) | \( T/K = 892 \) | \( x(\text{MoO}_3) = 0.6667 \) |
| \text{BaMo}_2\text{O}_7 \rightarrow \text{BaMoO}_4 + \text{Liquid} | Peritectic | \( x(\text{MoO}_3) = 0.75 \) | \( T/K = 917 \) | \( x(\text{MoO}_3) = 0.75 \) |
| \text{BaMo}_3\text{O}_{10} \rightarrow \text{Liquid} | Eutectic | \( x(\text{MoO}_3) = 0.822 \) | \( T/K = 893 \) | \( x(\text{MoO}_3) = 0.825 \) |

* The exact temperature or composition were not reported in the original publication, but read from the reported phase diagram.
Two measurements were performed on fresh material and averaged, yielding an enthalpy of transition equal to $\Delta_h^\text{eq}(\text{BaMo}_3\text{O}_{10}, \text{cr}, T_p) = (82.8\pm20.0)\text{kJ} \cdot \text{mol}^{-1}$. The results of the individual cycles are listed in Table 9. The reported uncertainties on the individual measurements are standard uncertainties. They include the uncertainty on the transition enthalpy of the reference material (Cs$_2$TeO$_4$) combined with the uncertainty associated with the choice of the baseline for the peak integration (linear, spline, or tangential sigmoid).

6. Thermodynamic modelling assessment

6.1. Optimized parameters

A summary of the optimized parameters for the Ba-Mo-O system is given in Table 10.

6.2. Thermodynamic data

The standard enthalpies of formation and standard entropies at 298.15 K optimized in this work are listed in Table 11. The values for BaMoO$_4$ are in good agreement with the selected values from the literature (see Section 2) considering the reported uncertainties. The optimized enthalpy of formation and standard entropy of BaMoO$_3$ are higher than recommended from the literature review, however. Such high values appeared necessary to match the decomposition temperature of this compound (calculated at $T = 1660 \text{K}$ and reported at $T = 1653 \text{K}$ by Paschoal et al. [66]). Since neither the enthalpy of formation nor the decomposition temperature for this compound are known with certitude, complementary measurements would be very valuable for the improvement of the thermodynamic model.

The calculated enthalpy increments and heat capacities of the same compounds are shown in Figs. 10a, 10 b, 11 a and 11 b. The agreement with the experimental data is very good. Note that the reported heat capacity equations are only valid above 298.15 K, and match the imposed constraints on $C_v^{\text{cr}}$(298.15 K), as selected from the literature review. In addition, the calculated transition enthalpy for the peritectic decomposition of BaMo$_3$O$_{10}$ is $\Delta_h^\text{m}(T_p) = 80.1 \text{kJ} \cdot \text{mol}^{-1}$, in very good agreement with the data measured in this work (i.e. $82.8\pm20.0 \text{kJ} \cdot \text{mol}^{-1}$). The transition enthalpy associated with the congruent melting of BaMoO$_4$ is $99.7 \text{kJ} \cdot \text{mol}^{-1}$, which is a reasonable value compared to the reported data for the isostructural CaMoO$_3$ ($102\pm6 \text{kJ} \cdot \text{mol}^{-1}$ [57]) and BaWO$_4$ ($96.9 \text{kJ} \cdot \text{mol}^{-1}$ [58]) compounds.

6.3. Phase diagram data

The calculated pseudo-binary section BaO-MoO$_3$ is shown in Fig. 12a and compared with the available phase diagram data.

The agreement with the data of Yanushkevich et al. [68] in the BaO-BaMoO$_4$ section, and the data measured in the BaMoO$_4$-MoO$_3$ section in this work and by Zhukovskii et al. [69], is generally good. The calculated invariant equilibria are listed in Table 12. Ba$_3$MoO$_6$ and BaMoO$_4$ melt congruently at $T = 1833 \text{K}$ and $T = 1734 \text{K}$, respectively. Ba$_3$MoO$_6$ undergoes a peritectic decomposition at $T = 1572 \text{K}$, and BaMoO$_3$$_{10}$ a peritectic decomposition at $T = 917 \text{K}$. BaMoO$_7$ is suggested to undergo a peritectoid decomposition at $T = 892 \text{K}$. The crystal structure of this compound remains unknown, however, and the post-XRD characterizations did not seem to indicate the presence of such a phase, by contrast with the measured calorimetric heat flow signals. Its existence thus still needs confirmation. The liquidus line between BaO and BaMoO$_4$ follows rather well the data of Yanushkevich et al. [68]. The situation on the BaMoO$_4$-MoO$_3$ section is more complex. The liquidus line is in good agreement with the results of Zhukovskii et al. [69] up to $x(\text{MoO}_3)\approx0.8$. In the MoO$_3$-rich region of the phase diagram, the liquidus line falls in between our data and that of Zhukovskii et al. [69] and Ustinov et al. [56]. Complementary measurements in the BaMoO$_4$-MoO$_3$ section, and more specifically of the liquidus line would be very valuable to ascertain the phase equilibria in this region and solve the discrepancy between the three sets of data.

Finally, isothermal sections calculated at $T = 300, 700, 1200$, and $1700 \text{K}$ are shown in Figs. 14a-d, respectively. The calculated equilibria ternary phase fields should be confirmed experimentally to ascertain the predictions of our model. The isothermal section computed at $T = 700 \text{K}$ is in rather good agreement with that of Dash et al. [53], although the comparison is not straightforward. Dash et al. included Ba$_3$Mo$_2$O$_{12}$, BaMoO$_4$, and Ba$_2$Mo$_2$O$_{10}$, which were not taken into account in the present model due to the lack of structural and thermodynamic data (see Section 2). Nevertheless, the computed ternary phase fields are similar, except for (Ba$_3$O$_4$)-BaMoO$_4$ calculated as a stable ternary phase field by Dash et al. [53], by contrast with the present work, showing instead an equilibrium between (Mo-BaO-B$_3$MoO$_3$) and (Mo-BaMo$_3$-Ba$_3$MoO$_6$).

7. Conclusions

The Ba-Mo-O system is rather complex, with a number of ternary barium molybdate phases reported in the literature. The thermodynamic modelling assessment for this system presented in this work includes BaMoO$_4$, Ba$_2$MoO$_6$, Ba$_3$MoO$_4$, BaMoO$_7$, BaMoO$_3$$_{10}$ and BaMoO$_3$ solid phases, for which sufficient structural information is available. To provide experimental input for the optimization of the model using the CALPHAD method, BaMoO$_4$, BaMoO$_3$, and BaMoO$_3$$_{10}$ have been synthesized, and further used for thermodynamic property measurements. The valence state of molybdenum in the former two compounds was confirmed to be $+VI$ and $+IV$, respectively, from XANES measurements at the Mo K-edge. Low-temperature heat capacity measurements on the same compositions have yielded $S_m^x$(BaMoO$_4$, cr, 298.15 K) = (160.4 ± 3.8) J · K$^{-1}$ · mol$^{-1}$ and $S_m^x$(BaMo$_3$, cr, 298.15 K) = (111.3 ± 2.8) J · K$^{-1}$ · mol$^{-1}$, respectively. Phase diagram measurements performed in the BaMoO$_4$-MoO$_3$ section have revealed discrepancies with respect to earlier studies. Finally, BaMoO$_3$$_{10}$ was suggested to undergo a peritectoid decomposition and the transition enthalpy associated with this invariant reaction was determined as $\Delta_h^\text{m}(\text{BaMoO}_3$$_{10}, \text{cr}, T_p) = (82.8\pm20.0)\text{kJ} \cdot \text{mol}^{-1}$. The optimized thermodynamic model shows overall a good agreement with the literature and newly measured data. This study has highlighted, however, a number of poorly known properties and unresolved issues:

- the thermodynamic properties (enthalpy of formation, entropy, transition enthalpies) of Ba$_3$MoO$_6$, Ba$_2$MoO$_6$, BaMoO$_7$, and BaMoO$_3$$_{10}$ have not been measured to this date;
- the existence of BaMoO$_7$ should be ascertained and its crystal structure resolved;
- the phase equilibria between BaMoO$_4$ and MoO$_3$ are still poorly known and uncertain, with quite large discrepancies between past literature studies and this work. The vapour pressures in the ternary phase fields and above pure compounds were not discussed, and will be the subject of future studies.

Declaration of Competing Interest

The authors report no declarations of interest.

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6 The stated uncertainty was derived based on the weighted average of the individual measurements.
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Appendix A. Calorimetry data in the BaMoO$_4$-MoO$_3$ pseudo-binary section

Fig. A.1. Heat flow curves measured in the BaO-MoO$_3$ phase diagram.

Fig. A.2. Heat flow curves measured in the BaO-MoO$_3$ phase diagram.

Appendix B. Thermodynamic modelling assessment of the Ba-O system
### Table B.1
Summary of the thermodynamic data for pure elements and oxides selected in the Ba-O system. SER refers to the phase of the element stable at 298.15 K.

| Phase | Gibbs energy (J·mol⁻¹) | Reference |
|-------|------------------------|-----------|
| Liquid (BaO)
(BaO₃, VaO₂, O) | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{VaO}}_2 - 2\Delta G^{\text{O}_2} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}_2}_{\text{gas}} - 993855 + 100.2257 \) | [26] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{VaO}}_2 - 2\Delta G^{\text{O}_2} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}_2}_{\text{gas}} - 993855 + 100.2257 \) | [79] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{VaO}}_2 - 2\Delta G^{\text{O}_2} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}_2}_{\text{gas}} - 993855 + 100.2257 \) | [79] |
| | \( \Delta G^{\text{VaO}}_2 - \Delta G^{\text{O}} = 2648.9 + 31.44T \) | [79] |
| | \( \Delta G^{\text{VaO}}_2 - \Delta G^{\text{O}} = 0 \) | [10] |
| | \( \Delta G^{\text{VaO}}_2 - \Delta G^{\text{O}} = 0 \) | [10] |
| | \( \Delta G^{\text{O}_2} = \) | [10] |
| Halite BaO
(BaO, O) | \( \Delta G^{\text{BaO}}_2 = -46278.5 + 370.1567 - 69.91257Tn(T) \) | [10] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [26] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [10] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [10] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [10] |
| | \( \Delta G^{\text{BaO}}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [10] |
| BaO₂ | \( G_{\text{BaO}} = -65504 \) | [26] |
| Gas
(BaOgasBa₂OgasO₂gasO₂) | \( \Delta G^{\text{BaO}_2}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [81] |
| | \( \Delta G^{\text{BaO}_2}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [81] |
| | \( \Delta G^{\text{BaO}_2}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [81] |
| | \( \Delta G^{\text{BaO}_2}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [81] |
| | \( \Delta G^{\text{BaO}_2}_2 = 2\Delta G^{\text{Ba}}_{\text{gas}} - 2\Delta G^{\text{O}} = 2G^{\text{Ba}}_{\text{gas}} + 2G^{\text{O}}_{\text{gas}} - 993855 + 100.2257 \) | [81] |

### Table B.2
Summary of the thermodynamic data for pure elements and oxides selected in the Ba-O system. SER refers to the phase of the element stable at 298.15 K.

| Phase | Gibbs energy (J·mol⁻¹) | Reference |
|-------|------------------------|-----------|
| G_{\text{Ba}}^{\text{SER}} = -178685.405 – 28.75190797 – 21.520686Tln(T) | [81] |
| \(+4.6163325 \cdot 10^{-4}\)T² – 1.0440133 \cdot 10^{-5}\)T – 6.531671T \(\cdot 10^{-4}\) (298 < T < 1000 K) | [-156541.305 – 187.1239187 – 52.09573433 (298 < T < 1000 K) | [81] |
| \(+0.1977270572 \cdot 2.34713833 \cdot 10^{-5}\)T² – 298.8954951T \(\cdot 10^{-5}\) (1000 < T < 1900 K) | | [81] |
| \(-444158.608 – 1453.693971 – 163.8061Tln(T) | | [81] |
| \(-0.0523775571 \cdot 2.155575 \cdot 10^{-5}\)T² – 6.89474507T \(\cdot 10^{-5}\) (1900 < T < 2600 K) | | [81] |
| \(+35485.605 – 1163.843257 – 129.623Tln(T) – 0.04764176T² | | [81] |
| \(+2.12269167 \cdot 10^{-5}\)T² – 30187.880T | | [81] |
| \(-50267.818 + 1933.17624T – 252.197Tln(T) + 0.02779515T² | | [81] |
| \(-6.7949667 \cdot 10^{-5}\)T² – 3.319067 \cdot 10^{-5}\)T² (3400 < T < 5300 K) | | [81] |
| \(+229718.753 – 82.414527T – 35.2043Tln(T) – 8.066315 \cdot 10^{-5}\)T² | | [81] |
| \(+2.7938433 \cdot 10^{-5}\)T² – 1.3210115T | | [81] |
| \(+34372.221 + 33.4913106T – 51.13698Tln(T) | | [81] |
| \(+0.32202522T² – 4.47438 \cdot 10^{-6}\)T² – 7.29072 – 67907.6T | | [81] |
| \(+10.35118359 – 158.369248T – 21.314586Tln(T) + 2.0490545 \cdot 10^{-5}\)T² | | [81] |
| \(+1.36280533 \cdot 10^{-6}\)T² – 1049988.5T | | [81] |
| \(+129260.91 – 2.0274389T – 34.435978Tln(T) | | [81] |
| \(-0.02453716T² – 3.34474 \cdot 10^{-5}\)T² – 126306.5T² | | [81] |
| \(-138831.073 – 79.887374T – 45.95129Tln(T) + 0.00391373T² | | [81] |

(continued on next page)
Table B.2 (continued)

| Phase | Gibbs energy/(J mol\(^{-1}\)) | Reference |
|-------|-----------------------------|-----------|
| \(G_{\text{BaO}}\) = -259922.912 + 87.0305009T + 55.15417T\(\ln(T)\) + 0.00662439T\(^2\) + 1.148982\(10^6\)T\(^3\) (298 < T < 1000 K) | [81] |
| \(G_{\text{BaO}}\) = -527525.33 + 137.41572T + 73.014T\(\ln(T)\) + 0.006415T\(^2\) + 1.054733\(10^6\)T\(^3\) (298 < T < 1000 K) | [25] |
| \(G_{\text{BaO}}\) = -263453.85 + 134.319936T - 62.3304T\(\ln(T)\) - 4.296195\(10^{-6}\)T\(^3\) + 5.31555 \(10^{-11}\)T\(^3\) + 632566.5T (900 < T < 3000 K) | |
| \(G_{\text{BaO}}\) = -255722167 + \(10^{-7}\)\(10^5\) + 1649835T\(^{-1}\) (1100 < T < 2400 K) | |
| \(G_{\text{BaO}}\) = -111463.795 + 19.5306837T - 39.56298T\(\ln(T)\) - 0.00507084T\(^2\) | |

Fig. B.1. Ba-O phase diagram computed at 1 bar, and comparison with the experimental data of Guntz and Benoit [100], Schumacher [101], Schriel [102], Foex [103], Kedrovskii et al. [104], Aptekar et al. [27], Mayorova et al. [28].

Fig. B.2. Heat capacity of BaO computed at 1 bar, and comparison with the experimental data of Cordfunke [105], Chekhovskoi [106], NIST-JANAF [107], Gmelin [108], and Anderson [109].

Fig. B.3. Enthalpy increment of BaO computed at 1 bar, and comparison with the experimental data of Cordfunke [105], Demirok [110], Irgashov [111] and Lander [112].

Fig. B.4. Heat capacity of BaO\(_2\) computed at 1 bar, and comparison with the experimental data of Wiedemann [113], Vedeneev [114], Herzfeld and Stiepel [115], and Gavrichev et al. [116].
Appendix C. Thermodynamic modelling assessment of the Ba-Mo system

Table C.1
Summary of the thermodynamic data for pure elements and oxides selected in the Ba-Mo system. SER refers to the phase of the element stable at 298.15 K.

| Phase | Gibbs energy/(J/mol⁻¹) | Reference |
|-------|------------------------|-----------|
| Liquid | \( \Delta G(Ba^{2+}) \times \text{Vox} \) | | [79] |
| (Ba\textsuperscript{2+},Mo\textsuperscript{4+})\times(Va\textsuperscript{O}) | | | |
| \( \Delta G(Mo^{4+}) \times \text{Vox} \) | | | [79] |
| \( \Delta G(Ba^{2+}) \times \text{Vox} \times 100000 \) | | | [10] |
| \( \Delta G(Mo^{4+}) \times \text{Vox} \times 50000 \) | | | [10] |
| (Ba,Mo) (bcc) | | | |
| (Ba,Mo)(Va\textsubscript{3}) | | | |
| \( \Delta G(Ba) \times 100000 \) | | | [79] |
| \( \Delta G(Mo) \times 50000 \) | | | [79] |
| (Ba\textsubscript{2},Mo\textsubscript{2},Mo\textsubscript{3}) | | | |
| Gas | | | |
| (Ba\textsubscript{2},Mo\textsubscript{2},Mo\textsubscript{3}) | | | |
| \( \Delta G_{\text{Ba}^{2+}} - \Delta H_{\text{Ba}^{2+}} + RT \ln(10^{10}P) \) | | | [81] |
| \( \Delta G_{\text{Mo}^{4+}} - 2\Delta H_{\text{Mo}^{4+}} + RT \ln(10^{10}P) \) | | | [81] |
| \( \Delta G_{\text{Mo}^{4+}} - \Delta H_{\text{Mo}^{4+}} + RT \ln(10^{10}P) \) | | | [79] |
| \( \Delta G_{\text{Mo}^{4+}} - 2\Delta H_{\text{Mo}^{4+}} + RT \ln(10^{10}P) \) | | | [79] |

(continued on next page)
Table C.1 (continued)

| Phase | Gibbs energy (J·mol⁻¹) | Reference |
|-------|-------------------------|-----------|
| Functions | \( G_{Mo}^{25} = -176865.405 - 28.7519079 T - 21.15208 Tln(T) \) | [79] |
| | +4.6163325 \(-10^3 T\) – 1.04400133 \(-10^3 T^2\) – 6531.6778 \(-10^3 T^3\) (289 < T/K < 1000 K) | |
| | \( 156541.305 + 187.123918 T - 52.90577 Tln(T) \) | |
| | +0.197727057 \(-2.37413383 \(-10^3 T\) – 2988594.57 \(-10^3 T^2\) (1000 < T/K < 1900 K) | |
| | \( 444158.608 + 1453.569391 T + 163.8061 Tln(T) \) | |
| | -0.052377557 \(-2.15595 T\) – 68947540.5 \(-10^4 T\) (1900 < T/K < 2600 K) | |
| | \( -558453.605 + 1163.843257 T - 129.0625 Tln(T) - 0.074674175 T \) | |
| | \(+2.12269167 \(-3.0187880 T\) (2600 < T/K < 3400 K) | |
| | -50267.818 \(-19.33.76244 T - 252.1979 Tln(T)\) + 0.027791557 T | |
| | \(-6.79449667 \(-10^3 T\) - 3.193667 \(-10^3 T\) (3400 < T/K < 5000 K) | |
| | \(+229718.753 + 82.4145257 - 55.20435 Tln(T)\) - 8.063615 \(-10^3 T\) | |
| | \(+2.79384333 + 10^3 T\) - 0.3120115 \(-10^3 T\) (5300 < T/K < 9600 K) | |
| \(G_{Mo}^{25} \) | \(G_{Mo}^{25} = +34.005 + 117.22478 \(-3.56414 Tln(T)\) | |
| | -0.003439369 \(-10^3 T\) \(-10^3 T\) + 2000 \(-10^2 T\) (289 < T < 5000 K) | |
| | +4.25419 \(-10^3 T\) (298 < T < 2896 K) | |
| | +1358.963 \(-271.66957 + 42.63828 Tln(T)\) (2896 < T < 5000 K) | |

Table C.2

Summary of the thermodynamic data for pure elements and oxides selected in the Ba-Mo system. SER refers to the phase of the element stable at 298.15 K.

| Phase | Gibbs energy (J·mol⁻¹) | Reference |
|-------|-------------------------|-----------|
| \(G_{Mo}^{25} \) | \(G_{Mo}^{25} = -176865.405 - 28.7519079 T - 21.15208 Tln(T) \) | [81] |
| | +4.6163325 \(-10^3 T\) – 1.04400133 \(-10^3 T^2\) – 6531.6778 \(-10^3 T^3\) (289 < T/K < 1000 K) | |
| | \( 156541.305 + 187.123918 T - 52.90577 Tln(T) \) | |
| | +0.197727057 \(-2.37413383 \(-10^3 T\) – 2988594.57 \(-10^3 T^2\) (1000 < T/K < 1900 K) | |
| | \( 444158.608 + 1453.569391 T + 163.8061 Tln(T) \) | |
| | -0.052377557 \(-2.15595 T\) – 68947540.5 \(-10^4 T\) (1900 < T/K < 2600 K) | |
| | \( -558453.605 + 1163.843257 T - 129.0625 Tln(T) - 0.074674175 T \) | |
| | \(+2.12269167 \(-3.0187880 T\) (2600 < T/K < 3400 K) | |
| | -50267.818 \(-19.33.76244 T - 252.1979 Tln(T)\) + 0.027791557 T | |
| | \(-6.79449667 \(-10^3 T\) - 3.193667 \(-10^3 T\) (3400 < T/K < 5000 K) | |
| | \(+229718.753 + 82.4145257 - 55.20435 Tln(T)\) - 8.063615 \(-10^3 T\) | |
| | \(+2.79384333 + 10^3 T\) - 0.3120115 \(-10^3 T\) (5300 < T/K < 9600 K) | |
| \(G_{Mo}^{25} \) | \(G_{Mo}^{25} = +34.005 + 117.22478 \(-3.56414 Tln(T)\) | |
| | -0.003439369 \(-10^3 T\) \(-10^3 T\) + 2000 \(-10^2 T\) (289 < T < 5000 K) | |
| | +4.25419 \(-10^3 T\) (298 < T < 2896 K) | |
| | +1358.963 \(-271.66957 + 42.63828 Tln(T)\) (2896 < T < 5000 K) | |

Appendix D. Supplementary data

Supplementary data associated with this article can be found, in the online version, at https://doi.org/10.1016/j.jeurceramsoc.2021.01.010.

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