

Effect of RF Magnetron Sputtered Nickel Oxide Thin Films as an Anode Buffer Layer in a P$_3$HT:PCBM Bulk Hetero-Junction Solar Cells

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Bulk heterojunction solar cells were investigated using poly(3-hexylthiophene) (P$_3$HT):[6,6]-phenyl-C$_{61}$ butyric acid methyl ester (PCBM) with a nickel oxide (NiO) anode buffer layer between the photoactive layer and an indium tin oxide (ITO) anode layer. The NiO anode buffer layer was deposited using radio frequency magnetron sputtering on an ITO electrode layer for effective hole transport and electron blocking. The NiO film is a $p$-type semiconductor with resistivity of $0.35 \ \Omega \ \text{cm.}$ The power conversion efficiency was improved substantially by the NiO anode buffer layer compared to a solar cell with an anode buffer layer made from poly(3,4-ethylenedioxythiophene) (PEDOT):poly(styrene sulfonate) (PSS). The solar cell with a 10 nm thick NiO anode buffer layer had a power conversion efficiency of 4.71%. These results are explained by the improved charge transport across the interface between the active layer and ITO electrode.

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1. Introduction

Bulk heterojunction (BHJ) solar cells based on organic materials are very attractive because of their low cost, large area, and light weight [1, 2]. For effective hole transport and electron blocking, PEDOT:PSS has commonly been used as an anode buffer layer (ABL) between an active layer of PCBM and an ITO electrode. However, PEDOT:PSS has problem with causing corrosion to the ITO, as well as non-uniform morphology and instability [3, 4]. Kawano et al. reported that PEDOT:PSS rapidly degrades the performance of a solar cell because of corrosion of the ITO anode electrode from the absorption of humidity in the air [5]. Therefore, it is necessary to develop a new ABL to replace the PEDOT:PSS in P$_3$HT:PCBM BHJ solar cells.

Suitable ABL materials should have sufficient optical transparency in the visible spectral region. They should also effectively transport holes and block electrons to the anode in a P$_3$HT:PCBM BHJ solar cell because P$_3$HT and PCBM are in direct contact with both electrodes. To achieve this, the conduction band energy ($E_c$) level of the ABL materials (which have $p$-type semiconductors) should be sufficiently higher than the lowest unoccupied molecular orbital (LUMO) energy level of the PCBM for electron protection. Furthermore, the valence band energy ($E_v$) level should be near the highest occupied molecular orbital (HOMO) energy of P$_3$HT for hole transport. There should also be ohmic contact at the interface between the P$_3$HT and ABL [6, 7].

NiO thin film has received considerable attention recently as a replacement for the PSS:PEDOT ABL in a P$_3$HT:PCBM BHJ solar cell NiO is a $p$-type transparent semiconductor with an $E_v$ level of $-5.4 \ \text{eV}$ and $E_c$ level of $-1.8 \ \text{eV}$. These properties result in effective hole transport and electron blocking [8]. Furthermore, NiO forms ohmic contact with P$_3$HT.

In the present study, we investigated the effect of a NiO ABL in a P$_3$HT:PCBM BHJ solar cell. A thin film of NiO was deposited by radio frequency (RF) magnetron sputtering. This method can be used to deposit films on a large area with good adhesion, high deposition, good thickness uniformity and high density [9]. The electrical and optical properties of the NiO film are strongly affected by the sputtering conditions, such as the Ar:O$_2$ gas flow ratios and thermal annealing temperatures.

2. Experimental

Corning glass (#1737) substrates were first cleaned using sequential ultrasonic baths in acetone, ethanol, and deionized water, followed by drying with an N$_2$ gun. Pre-sputtering was carried out for 5 min to eliminate contaminants on the target before deposition of the films. NiO films (80 nm) were then deposited on the substrates using RF magnetron sputtering (RF power: 100 W, distance between target and substrate: 10 cm, pressure: $10^{-3} \ \text{Torr}$, total gas flow rate: 100 sccm). The deposition was carried out using a target with a diameter of 2 in and various Ar:O$_2$ gas flow ratios (60:40, 50:50, 40:60, and 30:70) at room temperature in order to obtain optimal NiO films. The films were also deposited with a gas ratio of 30:70 and thermally annealed at 50, 100, 150, and 200$^\circ$C at low pressure ($10^{-3} \ \text{Torr}$, O$_2$ gas

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The electrical properties of the NiO films (80 nm) deposited at room temperature were measured as a function of the Ar:O\textsubscript{2} gas flow ratio. As the Ar:O\textsubscript{2} gas flow ratios increased, the resistivity and Hall mobility of the films gradually decreased and the carrier concentration of NiO films gradually increased, as shown in Fig. 1. The resistivity of the film deposited with a ratio of 30:70 was 0.95 Ω cm, which was the lowest value among the samples tested. As the partial oxygen pressure used in the deposition increases, the number of nickel vacancies should increase, and NiO\textsuperscript{2+} is substituted by Ni\textsuperscript{3+} in order to maintain the electro-neutrality, which resulted in a highly doped p-type semiconductor [10].

To decrease the resistivity and increase the transmittance of NiO films deposited with an Ar:O\textsubscript{2} flow ratio of 30:70, thermal annealing was carried out for 10 min at 50, 100, and 200 °C at low pressure (10\textsuperscript{−3} Torr, O\textsubscript{2} gas flow: 100 sccm). As the annealing temperature increased, the resistivity and Hall mobility of the NiO films gradually decreased, while the carrier concentration gradually increased, as shown in Fig. 2. The resistivity obtained with annealing at 200 °C was 0.35 Ω cm, which was the lowest value among the tested samples. These process conditions were the best for the NiO deposition in this experiment.
in Fig. 3. In addition, the $E_g$ values of NiO films decreased from 3.68 to 3.56 eV as the O$_2$ gas flow ratios increased. The $E_g$ values were calculated using the equation $\alpha h\nu = A(h\nu - E_g)^{1/2}$, where $A$ is the proportionality constant, $\alpha$ is the absorption coefficient, and $h\nu$ is the photon energy ($h$ — Plank’s constant, $\nu$ — frequency of the photon). The $E_g$ values were obtained by extrapolating the linear portion of the curve to zero absorption [11]. Based on the experimental results, we selected the Ar:O$_2$ flow ratio of 30:70 and thermal annealing at 200°C for 10 min at low pressure ($10^{-3}$ Torr, O$_2$ gas flow: 100 sccm) for the ABL. These conditions were optimal for the resistivity, optical transmittance, and $E_v$ level.

The NiO film obtained at 30:70 had a little lower optical transmittance than deposited with lower O$_2$ gas flow ratios (70:30 and 50:50). Nevertheless, the resistivity and $E_g$ value were also lower. When the $E_v$ level of the NiO film is lower than the HOMO energy level of P$_3$HT, the hole carriers generated in the P$_3$HT layer have an energy barrier between the P$_3$HT and NiO ABL. However, when the $E_v$ level of the NiO film approaches the HOMO energy level of P$_3$HT, the hole carriers are easier to transport from the P$_3$HT to the NiO ABL. This is possible when $E_g$ value of the NiO films decreases as the O$_2$ gas flow ratios increase in the NiO deposition process.

Figure 4 shows the UPS spectra (He(I) photon energy: 21.2 eV) of NiO films deposited with Ar:O$_2$ flow ratios of 50:50 and 30:70, and then thermally annealed at 200°C for 10 min. The $E_v$ level (–5.1 eV) of the NiO film deposited with a gas flow ratio of 30:70 was higher than that (–5.3 eV) obtained with a ratio of 50:50.

3.2. P$_3$HT:PCBM BHJ solar cell characterization

P$_3$HT:PCBM BHJ solar cells were fabricated with NiO ABLs (5, 10, and 15 nm) deposited with an Ar:O$_2$ flow ratio of 30:70 and thermal annealing at 200°C for 10 min. Solar cells were also fabricated using a PEDOT:PSS $p$-type ABL. The process conditions were the same except for the ABL process. Figure 5 shows the current density–voltage plots of the fabricated solar cells. The electrical parameters of the solar cells are summarized in Table I. For the solar cell fabricated with the PEDOT:PSS ABL, open-circuit voltage ($V_{oc}$), short-circuit current density ($J_{sc}$), fill factor (FF), and power conversion efficiency (PCE) were 0.69 V, 7.3 mA, 44%, and 3.03%, respectively. The $V_{oc}$, $J_{sc}$, and PCE values of the solar cell fabricated with 10 NiO ABL were 0.75 V, 14.3 mA, and 4.71%, respectively, which are higher than those obtained with the PEDOT:PSS ABL. Thus, the data indicate that replacing the PEDOT:PSS with the NiO ABL is feasible.

The $J_{sc}$ (14.3 mA/cm$^2$) value obtained with the 10 nm NiO ABL was much higher than previously reported values [12, 13].

The energy levels of the materials used are shown in Fig. 6. $p$-type NiO is generally known to have a wide band gap, with $E_f$ level of –5 eV and $E_v$ level of –5.4 eV [14]. The $E_v$ level of NiO film deposited with an Ar:O$_2$ gas flow ratio of 50:50 was –5.3 eV, while that
Fig. 5. Current density–voltage plots of \(P_3\)HT:PCBM BHJ solar cells fabricated with 5, 10, and 15 nm thickness NiO ABLs deposited using RF magnetron sputtering with an Ar:O\(_2\) gas flow ratio of 30:70 and thermally annealed at 200°C for 10 min, as well as PEDOT:PSS (100 nm) anode buffer layers.

Fig. 6. Energy level diagram of a \(P_3\)HT:PCBM BHJ solar cell with a NiO anode buffer layer.

Obtained at 30:70 was -5.1 eV as shown in Fig. 4. The \(E_v\) levels increased with the O\(_2\) flow ratios. The \(E_v\) level of the NiO ABL obtained with a ratio of 30:70 was near the HOMO energy level (≈ 0.1 below \(E_f\)) of \(P_3\)HT, as shown in Fig. 4. When \(E_v\) level of the NiO ABL is close to the HOMO energy level of \(P_3\)HT, the holes generated at the HOMO energy level of \(P_3\)HT are transported more easily to the \(E_v\) of the NiO ABL. The LUMO energy level (≈ -1.8 eV) of NiO is higher than that of PCBM (≈ -4.0 eV), as shown in Fig. 6 [14]. Thus, the transport of electrons generated from the PCBM to the ITO electrode would be prevented with the NiO ABL.

The solar cell fabricated with a 10 nm NiO ABL had a much higher \(J_{sc}\) (14.3 mA/cm\(^2\)) than those of the other samples 4.65 mA/cm\(^2\) (5 nm) and 9.18 mA/cm\(^2\) (15 nm). However, FF for the 10 nm AB (44%) was lower than the other samples of 50% (5 nm) and 53% (15 nm). PCE obtained with the 10 nm NiO ABL (4.71%) was higher than those obtained with thickness of 5 (1.9%) and 15 nm (3.96%). When the NiO film thickness is 5 nm, the layer might not completely cover the ITO substrate. Therefore, the parallel resistance (420 \(\Omega\)) was lower for this thickness than for the 10 nm (2996 \(\Omega\)) and 15 nm NiO ABLs (17476 \(\Omega\)), which fully covered the ITO substrate, as shown in Table I. The PCE is very low when the NiO film thickness is less than 8 nm, which is attributed to the large leakage current and insufficient electron blocking [15]. The PCE of solar cell obtained with the 10 nm NiO ABL was higher than that obtained with the 15 nm NiO ABL. This was probably due to the difference in series resistance between the 10 nm (388 \(\Omega\)) and 15 nm (655 \(\Omega\)) layers.

Table I

| Film thickness [nm] | \(V_{OC}\) [V] | \(J_{sc}\) [mA/cm\(^2\)] | FF [%] | PCE [%] | \(R_S\) [\(\Omega\)] | \(R_{Sh}\) [\(\Omega\)] |
|--------------------|----------------|-----------------|-------|--------|----------------|----------------|
| PEDOT: PSS (100)   | 0.69           | 7.3             | 44    | 3.03   | 246           | 2330          |
| NiO (5)            | 0.82           | 4.65            | 50    | 1.90   | 1008          | 420           |
| NiO (10)           | 0.75           | 14.3            | 44    | 4.71   | 388           | 2996          |
| NiO (15)           | 0.82           | 9.18            | 53    | 3.96   | 655           | 17476         |

4. Conclusions

We have investigated NiO thin films obtained using RF magnetron sputtering with different gas flow ratios of Ar:O\(_2\) and thermal annealing temperatures to replace PEDOT:PSS ABLs in \(P_3\)HT: PCBM BHJ solar cells. PCE was much improved when using the 1 nm thickness NiO ABL (471%) compared to that obtained using the PEDOT:PSS ABL (3.03%). For the 10 nm NiO ABL obtained with an gas flow ratio of 30:70 and thermally annealed at 200°C for 10 min, the \(E_v\) level was close to the HOMO energy level (≈ 0.1 below of \(E_f\)) of \(P_3\)HT. Therefore, the holes generated at the HOMO energy level of \(P_3\)HT would transport easily to \(E_v\) of the NiO ABL. These results show that the NiO ABL could replace the PEDOT:PSS ABL in \(P_3\)HT: PCBM BHJ solar cells.
References

[1] H.-L. Yip, S.K. Hau, N.S. Baek, A.K.-Y. Jen, *Appl. Phys. Lett.* **92**, 193313-1 (2008).

[2] B.C. Thompson, J.M.J. Frechet, *Chem. Int. Ed.* **47**, 58 (2008).

[3] H.-K. Park, J.-W. Kang, S.-I. Na, D.-Y. Kim, H.-K. Kim, *Sol. Energy Mater. Sol. Cells* **93**, 1994 (2009).

[4] F.C. Krebs, T. Tromholt, M. Jorgensen, *Nanoscale* **2**, 873 (2010).

[5] K. Kawano, R. Pacios, D. Poplavskyy, J. Nelson, D.D.C. Bradley, J.R. Durrant, *Sol. Energy Mater. Sol. Cells* **90**, 3520 (2000).

[6] V. Shrotriya, G. Li, Y. Yao, C.-W. Chu, Y. Yang, *Appl. Phys. Lett.* **88**, 073508 (2006).

[7] M.S. White, D.C. Olson, S.E. Shaheen, N. Kopidakis, D.S. Ginley, *Appl. Phys. Lett.* **89**, 143517 (2006).

[8] K.X. Steirer, J.P. Chesin, N.E. Widjonarko, J.J. Berry, A. Miedaner, D.S. Ginley, D.C. Olson, *Org. Electr.* **11**, 1414 (2010).

[9] K. Ellmer, *J. Phys. D Appl. Phys.* **33**, R17 (2000).

[10] S. Seo, M.J. Lee, D.H. Seo, E.J. Jeoung, D.-S. Suh, Y.S. Joung, I.K. Yoo, I.R. Hwang, S.H. Kim, I.S. Byun, J.-S. Choi, B.H. Park, *Appl. Phys. Lett.* **85**, 5655 (2004).

[11] J. Tauc, R. Grigorovici, A. Vancu, *Phys. Status Solidi B* **15**, 627 (1966).

[12] R. Betancur, M. Maymo, X. Elias, L.T. Vuong, J. Martorell, *Sol. Energy Mater. Sol. Cells* **95**, 735 (2011).

[13] D.T. Nguyen, A. Ferrec, J. Keraudy, J.C. Bernede, N. Stephan, L. Cattin, P.-Y. Jouan, *Appl. Surf. Sci.* **311**, 110 (2014).

[14] N. Sun, G. Fang, P. Qin, Q. Zheng, M. Wang, X. Fan, F. Cheng, J. Wan, X. Zhao, *Sol. Energy Mater. Sol. Cells* **94**, 2328 (2010).

[15] N. Sun, G. Fang, P. Qin, Q. Zheng, M. Wang, X. Fan, F. Cheng, J. Wan, X. Zhao, *Sol. Energy Mater. Sol. Cells* **94**, 2328 (2010).