Kinetic modelling of intraband carrier relaxation in bulk and nanocrystalline lead-halide perovskites

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The relaxation of high-energy “hot” carriers in semiconductors is known to involve the redistribution of energy between hot and cold carriers, as well as the transfer of energy from hot carriers to phonons. Over the past few years, these two processes have been identified in lead-halide perovskites (LHPs) using ultrafast pump–probe experiments, but their interplay is not fully understood. Here we present a practical and intuitive kinetic model that accounts for the effects of both hot and cold carriers on carrier relaxation in LHPs. We apply this model to describe the dynamics of hot carriers in bulk and nanocrystalline CsPbBr₃ as observed by multi-pulse “pump–push–probe” spectroscopy. The model captures the slowing of the relaxation dynamics in the materials as the number of hot carriers increases, which has previously been explained by a “hot-phonon bottleneck” mechanism. The model also correctly predicts an acceleration of the relaxation kinetics as the number of cold carriers in the samples is increased. Using a series of natural approximations, we reduce our model to a simple form containing terms for the carrier–carrier and carrier–phonon interactions. The model can be instrumental for evaluating the details of carrier relaxation and carrier–phonon couplings in LHPs and other soft optoelectronic materials.

1. Introduction

Lead-halide perovskites (LHPs) are rapidly emerging as a promising class of semiconducting material for future optoelectronic applications. Since the first report of 3.8% power conversion efficiency (PCE) in 2009,¹ laboratory-scale LHP solar cells can now routinely achieve PCEs above 20%, with a record PCE over 25%.² Compared to the top PCEs of other third-generation solution-processable photovoltaic materials, this value is impressively close to the ~33% PCE given by the theoretical “Shockley–Queisser” limit for single-junction solar cells.³ A key factor underpinning this limit is the rapid dissipation of energy from “hot” carriers following the photoexcitation of the semiconductor above its bandgap. Early work by Ross and Nozik postulated that the performance of a solar cell could be dramatically enhanced if the hot carriers could be utilised.⁴ The efficiency of this utilisation depends on how quickly the hot carriers dissipate their excess energy, reflected by the relaxation rate.

The intraband relaxation rate depends on many factors which control the carrier dynamics, including the interaction between carriers (i.e. redistribution of heat between the carriers, called thermalisation), and the interactions between carriers and phonons (i.e. the removal of excess energy through lattice vibrations, called cooling).⁵ Cooling has been studied extensively in LHPs using pump–probe methods. A common finding is a slowdown of the cooling kinetics with increasing carrier density.⁶–¹² This phenomenon has also been observed in traditional semiconductors such as GaAs,¹³,¹⁴ and is often referred to as the “hot-phonon bottleneck”. For LHPs, this behaviour is thought to occur when polarons (i.e. localised distortions of the lattice surrounding a carrier) spatially overlap, leading to the reabsorption of hot phonons by adjacent carriers.¹⁵ We previously demonstrated that this behaviour in LHPs is sensitive to the material composition, and is particularly pronounced in the all-inorganic CsPbBr₃ compared to its hybrid counterparts with vibrationally active organic cations.¹⁶

Most commercially available transient absorption methods do not have the prerequisite time resolution to observe
thermalisation, but high-resolution two-dimensional electronic spectroscopy has been employed to demonstrate this effect in LHPs.\textsuperscript{17,18} These studies show that the high-energy carriers produced by above-gap excitation reach a thermal distribution through carrier scattering at the early stage of relaxation \((<100\text{ fs})\), followed by the carrier–phonon interactions where the heat is dissipated in the lattice \((\sim 500\text{ fs})\). These reports provide valuable mechanistic insights, but the methods lack the ability to distinguish the roles of the hot and cold carriers on carrier relaxation.

Here, we propose a practical kinetic model to describe the effects of the hot and cold carrier populations on the carrier relaxation kinetics in CsPbBr\textsubscript{3} bulk and nanocrystal (NC) films, and apply this model to explain the results of “pump–push–probe” measurements.\textsuperscript{19} Our results convey two competing pathways for the dissipation of the excess hot carrier energy into either optical phonons or cold carriers. By applying simple approximations, we reduce this model into a single intuitive equation containing terms for the carrier–carrier and carrier–phonon interactions. At sufficiently high cold carrier concentration, the interactions between hot and cold carriers dominate over the hot-phonon bottleneck behaviour, leading to accelerated carrier relaxation.

2. Experimental methods

2.1 Sample preparation

Ca\textsubscript{2} substrates (12 mm diameter, 1 mm thickness, EKSMA Optics) were first treated with O\textsubscript{2} plasma to improve wetting.

The bulk CsPbBr\textsubscript{3} films were prepared by dissolving 1:1 mole ratio of PbBr\textsubscript{2} (Alfa Aesar, 99.9%) and CsBr (Alfa Aesar, 99.9%) in dimethyl sulfoxide (Sigma-Aldrich, 99.9%) with concentration of 0.4 M and stirring at 50 °C overnight. Afterwards, \sim 35 \mu L of the as-prepared solution was added onto the substrate and spin-coated at 3000 rpm for 2 min. The solvent extraction method was then applied by using 300 \mu L chloroform after 1 min spinning.\textsuperscript{20,21}

The as-formed thin films were then thermally annealed on a hot plate at 100 °C for 10 min inside the glovebox.

The CsPbBr\textsubscript{3} NCs were synthesised according to a previously reported method.\textsuperscript{22} 1 mL lead-oleate \((0.5 \text{ M in 1-octadiene (ODE))}, 0.8 \text{ mL Cs-oleate (0.4 M in ODE), 0.0445 g of N,N-(dimethyloctadecylammonio)-propanesulfonate and 10 mL ODE were added to a 25 mL three-neck flask and heated to 130 °C under vacuum. As soon as the reaction temperature was reached, the atmosphere was changed to nitrogen and 0.42 g of oleylamine bromide, dissolved in 3 mL of toluene, was injected. The reaction mixture was immediately cooled to room temperature by means of an ice bath.

The crude solution was then centrifuged at 29 500 g for 10 min. The precipitate was discarded and ethylacetate (30 mL) was added to the supernatant \((ca. 15 \text{ mL})\). The mixture was centrifuged at 29 500 g for 10 min and the precipitate was dispersed in toluene \((3 \text{ mL})\). This solution was subjected to three additional rounds of precipitation with ethylacetate \((6 \text{ mL})\) and centrifugation at 29 500 g for 1 min and subsequent redispersion using toluene \((3 \text{ mL})\). All films were determined to be \sim 300 nm thick by UV-vis ellipsometry.

2.2 UV-vis spectroscopy

In order to determine the carrier density induced by the pump, the absorption cross section of the samples was calculated from the linear absorption spectra. These spectra were obtained from a UV-vis spectrometer (Shimadzu 2600, equipped with an ISR-2600Plus integrating sphere). A 1 nm sampling interval was used with a 5 nm slit width.

2.3 Ultrafast infrared differential transmission measurements

A Ti:sapphire regenerative amplifier (Astrella, Coherent) was used to pump two optical parametric amplifiers (TOPAS-Prime, Coherent) with 1.55 eV pulses at a repetition rate of 4 kHz and pulse duration of \sim 35 fs. The signal output \((\sim 0.95 \text{ eV})\) of one of the optical parametric amplifiers was directed into a \beta-barium borate crystal (EKSMA Optics) along with the residual part of the fundamental \((1.55 \text{ eV})\) from the regenerative amplifier. Sum frequency generation yielded the 2.5 eV \sim 60 fs pump pulse. The idler output \((0.6 \text{ eV}, \sim 50 \text{ fs})\) of the other optical parametric amplifier was split in a 9:1 ratio. The more intense portion was used as the push, and the weaker portion was used as the probe. The pump and probe beams were collinearly overlapped and focused onto a \sim 0.2 mm diameter spot on the sample. The push was focused onto the same spot less tightly \((0.4 \text{ mm})\) in a non-collinear \((\sim 5°)\) geometry with respect to the pump/probe. Beam diameters were confirmed with pinholes. The transmission of the probe through the sample was recorded with an amplified PbSe photodetector (PDA20H-EC, ThorLabs). This was connected to a lock-in amplifier (SR-830, Stanford Research Systems) coupled to a chopper in the pump beam. The chopper operated at half the repetition rate of the amplifier to block every other pump pulse. The delay between the pump, push and probe beams were controlled with mechanical delay stages. The sample was measured in a nitrogen-purged quartz cuvette.

3. Experimental results

The linear absorption spectra of the bulk and NC samples are shown in Fig. 1(a). As expected, the NC sample exhibits a slightly blueshifted absorption onset due to weak quantum confinement.\textsuperscript{23} The NCs studied here have a cubic shape and an average edge-length of 8 \pm 2 nm, as confirmed by the electron micrograph in the inset of Fig. 1(a). A more lengthy discussion of the material properties of the two samples is given elsewhere.\textsuperscript{19} Taking together the weak confinement and ordered packing of the NCs, we anticipate that carriers can move within and between NCs. The high mobility of carriers in films of similarly sized CsPbBr\textsubscript{3} NCs was recently demonstrated by optical pump-terahertz probe experiments.\textsuperscript{24}

To investigate the intraband relaxation dynamics in the samples, we used a “visible pump–IR push–IR probe” pulse sequence, which has been outlined in previous works.\textsuperscript{16,19,25,26} and recently adapted to study hot carrier extraction in LHPs.\textsuperscript{27} As depicted in Fig. 1, a 2.5 eV pump generates excited carriers...
which promptly relax to the band edges. These free carriers are monitored by the differential transmission of a near-IR probe at 0.6 eV, which corresponds to an intraband absorption in LHPs.\textsuperscript{28–30} After a fixed delay (12 ps), an intense 0.6 eV push pulse converts the cold carriers into hot carriers, which is observed as a bleaching of the differential transmission signal. As the hot carriers relax, the pump–probe signal recovers. We fit the recovery of the probe signal with a monoexponential curve to extract the relaxation time, \(\tau_{\text{relax}}\). The intensity of the pump and push pulses are used to control the cold and hot carrier densities, respectively. The representative fluence-dependent pump–probe decay kinetics in Fig. S1 (ESI)\textsuperscript{†} are not purely exponential, which is suggestive of free carrier as opposed to monomolecular exciton recombination.

The total number of (hot and cold) carriers per unit volume in the system, \(n_{\text{exc}}\), is calculated using eqn (1). \(E_{\text{inc}}\) is the incident energy per pump pulse, \(A\) is the absorbance of the sample at the pump photon energy \(E_{\text{exc}}\), \(R\) is the beam radius and \(d\) is the sample thickness.

\[
n_{\text{exc}} = \frac{E_{\text{inc}}(1 - 10^{-4})}{\pi R^2 d E_{\text{exc}}} \quad (1)
\]

The initial hot carrier density, \(n_{\text{hot}}\), can be obtained from the product of \(n_{\text{exc}}\) and the ratio between the amplitudes of bleach and the signal before the arrival of the push pulse, labelled II and I in Fig. 1(d). The dependence of \(\tau_{\text{relax}}\) on the initial density of the hot and cold carriers is outlined in Fig. 2. For a given value of \(n_{\text{cold}}^0\) (cold carrier density just before the push), we find that \(\tau_{\text{relax}}\) increases with \(n_{\text{hot}}^0\). These dynamics agree with the literature and our own previous findings which invoke the hot-phonon bottleneck effect.\textsuperscript{16,19} Interestingly, we also find that \(\tau_{\text{relax}}\) tends to decrease with increasing \(n_{\text{exc}}\), suggesting that the hot-phonon bottleneck plays a lesser role in intraband relaxation when there are many cold carriers in the system. From this, we can deduce that the carrier relaxation occurs via at least two competing pathways. When there are few cold carriers, carrier–phonon interactions dominate over the hot-cold carrier interactions, and the hot-phonon bottleneck primarily governs the relaxation dynamics. At higher cold carrier density, relaxation via carrier–carrier interactions starts to dominate, resulting in the overall faster relaxation and the reduced impact of the hot-phonon bottleneck.\textsuperscript{19}

4. Kinetic modelling

We interpret our experimental data as the relaxation of hot carriers in the LHP samples through carrier–carrier and/or carrier–phonon interactions. To gain more insight into the interplay between these processes, we have developed a numerical model describing each process that is eventually reduced to a single equation, shown later. To fully describe the data, we take the following approximations:

- For simplicity, and based on the small binding energies and comparable effective masses of electrons and holes,\textsuperscript{31} we ignore exciton effects and assume that electrons and holes exhibit similar behaviour;
- Carriers can be either hot or cold, hence the sum of the number of hot and cold carriers (\(n_{\text{hot}}\) and \(n_{\text{cold}}\)) is equal to the total number of carriers (\(n_{\text{exc}}\));
- Here, hot carriers have an excess energy of \(\sim 0.6\) eV, comparable to the push energy. Just before the push arrives, \(n_{\text{hot}}\) is negligible;
- Carrier relaxation involves two co-existing pathways, each with characteristic rates; (i) carrier–carrier interaction, when a hot carrier scatters with other hot or cold carriers and loses (some of) its excess energy; and (ii) carrier–phonon interaction,
when the excess energy is transferred into an unoccupied phonon mode. The phonon mode becomes occupied, and the carrier relaxes;

- Total number of phonon modes \(N_{\text{ph}}\) is a constant for the material. The modes can be either occupied or unoccupied.

All simulations are presented for the unit volume (cm\(^{-3}\)). The initial system of equations is shown in eqn (2) with the initial conditions shown in eqn (3):

\[
\begin{align*}
\frac{dn_{\text{hot}}}{dt} &= \dot{I}_{\text{push}} n_{\text{hot}} - 2n_{\text{exc}} n_{\text{hot}} - \beta n_{\text{ph}} n_{\text{hot}} \\
\frac{dn_{\text{cold}}}{dt} &= - \frac{dn_{\text{hot}}}{dt} - k_{\text{rec}} n_{\text{cold}} \\
\frac{dn_{\text{ph}}}{dt} &= - \beta n_{\text{ph}} n_{\text{hot}} + \nu (N_{\text{ph}} - n_{\text{ph}})
\end{align*}
\]

(2)

\[
\begin{align*}
n_{\text{hot}} &= 0 \\
n_{\text{cold}} &= n_{\text{cold}}^0 \\
n_{\text{ph}} &= N_{\text{ph}}
\end{align*}
\]

(3)

where \(n_{\text{hot}}, n_{\text{cold}}\) and \(n_{\text{exc}}\) are the number of hot, cold and total number of carriers, respectively; \(N_{\text{ph}}\) and \(n_{\text{ph}}\) are the total number of phonon modes and the number of unoccupied phonon modes, respectively; \(n_{\text{cold}}^0\) is the cold carrier density at \(t = 0\) (just before the push pulse arrives); \(\dot{I}_{\text{push}}\) is the number of photons absorbed from the push pulse per unit time, expressed as a time-dependent Gaussian envelope function; \(\beta\) is the rate constant for the carrier-carrier interaction; \(\nu\) in the rate constant for carrier-phonon interaction; \(k_{\text{rec}}\) is the rate constant for the recombinination of carriers; and \(\nu\) is the rate constant for phonon vacancy freeing.

The hot-phonon bottleneck arises from the \(- \beta n_{\text{ph}} n_{\text{hot}}\) component of the rate equation for \(n_{\text{hot}}\); this component is proportional to the unoccupied phonon density, which depletes during the carrier-phonon interaction. The bottleneck becomes prominent if the rate of phonon depletion is greater than or comparable to the rate of phonon scattering (the dissipation of hot phonons to the lattice), i.e., \(- \beta n_{\text{hot}} n_{\text{ph}} \geq \nu (N_{\text{ph}} - n_{\text{ph}})\). The effect of carrier density is expressed in \(- 2n_{\text{exc}} n_{\text{hot}}\); an increased rate of carrier–carrier interactions is expected for a greater number of carriers. When carrier relaxation is modelled without this term, \(\tau_{\text{relax}}\) increases exponentially with respect to \(n_{\text{hot}}\). This is clearly not what we observe in the experimental data in Fig. 2. Fig. S2 (ESI†) shows that a similar issue arises when \(- 2n_{\text{exc}}\) is substituted for \(- 2n_{\text{cold}}\), where \(n_{\text{cold}}\) is obtained from \(n_{\text{exc}} - n_{\text{hot}}\). These findings corroborate the occurrence of strong carrier–carrier interactions when carriers are elevated to the hot state and a significant proportion of the cold carriers are depleted by the push pulse. Based on this, we are confident that we can neglect the absolute temperature of the carriers in our model. In any case, the sub-ps hot carrier lifetimes we observe are consistent with other pump-probe studies that use a Maxwell–Boltzmann temperature model to describe a Fermi–Dirac distribution of carriers.6–12

This system can be numerically solved to provide a full picture of the process, as shown in Fig. 3(a). The result is comprehensive but involves too many parameters for the analysis of carrier dynamics. Considering that the \(\sim 50\) fs push pulse is shorter than the \(\sim 500\) fs cooling times,6,19 we can assume that the intraband excitation is immediate and hence \(\dot{I}_{\text{push}} P_{\text{cold}} \approx 0\) after 100 fs, i.e. the time window of interest. Furthermore, the recombination lifetime is relatively slow \((>100\) ps\)),32 and thus can be neglected \((k_{\text{rec}} \approx 0)\). Consequently, \(n_{\text{exc}}\) becomes a constant \((n_{\text{exc}} = n_{\text{exc}}^0)\) and the rate equation for \(n_{\text{cold}}\) becomes obsolete.

The system simplifies into the form shown in eqn (4), with the initial conditions shown in eqn (5):

\[
\begin{align*}
\frac{dn_{\text{hot}}}{dt} &= - 2n_{\text{exc}} n_{\text{hot}} - \beta n_{\text{ph}} n_{\text{hot}} \\
\frac{dn_{\text{ph}}}{dt} &= - \beta n_{\text{ph}} n_{\text{hot}} + \nu (N_{\text{ph}} - n_{\text{ph}})
\end{align*}
\]

(4)
which simplifies to:

\[
\begin{align*}
    n_{ph} = N_{ph} \exp \left( \frac{\beta}{2n_{exc}} (\Delta n_{hot} - \Delta n_{hot}) \right)
\end{align*}
\] (8)

where \( n_{hot} = n_{hot} - n_{0} \) and \( n_{ph} = n_{ph} - n_{ph} \).

In order to derive the equation for further simplification, and extract a physical quantity related to polaron overlap, we assume that each hot carrier occupies a specific volume, \( V' \), where carrier–phonon coupling can occur. Then we divide a unit volume \( V \) of the lattice into uniform cells with the volume \( V' \). The hot carriers can occupy any of the cells, and can either overlap or not overlap with each other. The phonons are assumed to be evenly scattered across the lattice, and only one of the overlapping hot carriers can interact with the phonon modes in the cell. When \( n_{hot} - n_{0} \) \( V' \) carriers are coupled and dropped into the cells with the volume \( V' \) in a lattice with \( N_{ph} \) phonons per unit volume, then the average number of accessible phonons per unit volume \( (n_{ph}) \), can be expressed by eqn (9):

\[
\begin{align*}
    n_{ph} &= N_{ph} \left( 1 - V' \left( n_{hot} - n_{0} \right) V' \right)
\end{align*}
\] (9)

Given that \( \frac{V'}{V} \) is a small number, the following approximation can be made:

\[
\begin{align*}
    n_{ph} &= N_{ph} \exp (V' n_{hot})
\end{align*}
\] (10)

Fig. 4(a) and (b) compare the expression of \( n_{ph} \) from the eqn (4), (5) and (10) in the time domain. We demonstrate that the use of eqn (10) is a reasonable approximation of \( n_{ph} \) in eqn (4) and (5), especially at early times < 1 ps.

The rate equation for \( n_{hot} \) can be expressed as:

\[
\begin{align*}
    \frac{dn_{hot}}{dt} &= -2n_{exc}n_{hot} - \beta n_{ph} \exp (-V' \Delta n_{hot}) n_{hot}
\end{align*}
\] (11)

where \( \alpha, \beta, V' \) and \( N_{ph} \) are constants. In this expression, \( \alpha \) is the rate constant for the carrier–carrier interaction, \( \beta \) is the rate constant for the carrier–phonon interaction, \( V' \) is the carrier–phonon interaction volume and \( N_{ph} \) is the number of unoccupied phonon modes per unit volume in the lattice.

Eqn (11) includes the product of two arbitrary constants \( (\beta N_{ph}) \), allowing the elimination of a parameter. \( \beta \) is defined as the average spatial volume around a hot carrier where a phonon mode will interact with the carrier with \( \sim 100\% \) probability in 1 ps. \( \beta N_{ph} \) can be substituted by \( \rho V' \), where \( V' \) is the volume of space around a hot carrier where phonons are coupled, and \( \rho \) is the number of carrier–phonon interactions per unit volume per 1 ps. Both \( \beta N_{ph} \) and \( \rho V' \) represent the number of carrier–phonon interactions per hot carrier per 1 ps. Hence, the rate equation can be expressed as:

\[
\begin{align*}
    \frac{dn_{hot}}{dt} &= -2n_{exc}n_{hot} - \rho V' \exp (-V' \Delta n_{hot}) n_{hot}
\end{align*}
\] (12)

The hot-phonon bottleneck arises from the reduced number of available phonon modes with increasing hot carrier density, expressed as \( \rho V' \exp (-V' \Delta n_{hot}) \). The carrier–carrier interaction is contained in the –2n_{exc}n_{hot} component; the rate of the interaction is proportional to the carrier density. The relaxation time is calculated using the following expression:

\[
\begin{align*}
    \tau_{relax} &= \int_{n_{hot}}^{n_{hot}} \frac{dt}{n_{hot}}
\end{align*}
\] (13)

Based on this equation, we fit the experimental data from the CsPbBr₃ NCs and bulk material for comparison. The fits and experimental data are shown in Fig. 2 and 5 for the respective samples. It should be emphasised that all data are globally fit with a handful of free parameters, and slight...
The predicted onset of a Mott transition (overlap of polarons), taking this, and the fact that the carrier delocalisation distance in voids which are substantially smaller than the NCs themselves.

Table 1 Material constants for the, carrier–carrier interaction (\(a\)), carrier–phonon interaction (\(\rho\)) and the carrier–phonon interaction volume (\(V'\)) derived from the fit.

| Material          | \(a\) (10\(^{-18}\) cm\(^3\) ps\(^{-1}\)) | \(\rho\) (10\(^{-18}\) cm\(^3\) ps\(^{-1}\)) | \(V'\) (10\(^{-18}\) cm\(^3\)) |
|-------------------|-----------------------------------------|------------------------------------|-----------------|
| CsPbBr\(_3\) NCs  | 1.08                                    | 0.43                               | 7.00            |
| Bulk CsPbBr\(_3\)| 1.95                                    | 0.46                               | 7.58            |

Fig. 5 Plot of experimental data (points) and fits from the model (solid lines) for bulk CsPbBr\(_3\).

Phonons in each of the materials; in other words how far the lattice will “feel” the presence of a charge. We extract values of 12.1 and 11.9 nm for bulk CsPbBr\(_3\) and the CsPbBr\(_3\) NCs respectively. The fact that these values exceed the average NC edge implies that there is very efficient exchange of electronic and vibrational energy in LHP NC solids, which takes place on the sub-ps timescale. Given this timescale, we can use the model parameters to compare the carrier–carrier and carrier–phonon interaction distances. The values above indicate that for CsPbBr\(_3\) and its NC analogue, the typical distance at which carrier–phonon interactions start playing a role is \(~ 12\) nm. This is slightly larger than the typical carrier–carrier interaction distance which, when deduced from \(a\), is \(~ 7\) nm. These absolute values are higher estimates of the actual numbers due to the anticipated underestimation of the local carrier density. However, the estimated \(a\) and \(V'\) depend on carrier density in the same way, so the ratio between these parameters should be captured by the model correctly. Our observation that the carrier–carrier interactions are active at shorter distances than the carrier–phonon interactions is consistent with the notion of polaronic screening in LHPs.\(^{39-41}\)

5. Conclusion

In summary, we developed a numerical model which includes “carrier–carrier” and “carrier–phonon” interaction terms to describe the role of the hot and cold carriers on the carrier relaxation kinetics of bulk and nanocrystalline CsPbBr\(_3\) films determined by “pump–push–probe” spectroscopy. The observation of slower relaxation kinetics at higher pump intensity (higher hot carrier density) is explained by invoking a hot-phonon bottleneck mechanism where overlapping hot polarons compete for phonons in order to relax. This is incorporated in our model by the carrier–phonon term which describes the spatial overlap between the hot polaron states. Meanwhile, the observation of faster relaxation for higher pump intensity is attributed to the redistribution of energy from a hot carrier to adjacent cold carriers, as described by the carrier–carrier term. By fitting our model to the experimental results, we demonstrate the interplay between these contrasting effects, and propose a way to estimate the phonon density of the lattice and the rate constant for carrier–carrier and carrier–phonon interactions.

Conflicts of interest

There are no conflicts to declare.

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References
1 A. Kojima, K. Teshima, Y. Shirai and T. Miyasaka, J. Am. Chem. Soc., 2009, 131, 6050–6051.
2 NREL, Best Research-Cell Efficiencies Chart, https://www.nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.20190802.pdf, accessed 30 June 2020.
3 W. Shockley and H. J. Queisser, J. Appl. Phys., 1961, 32, 510–519.
4 R. T. Ross and A. J. Nozik, J. Appl. Phys., 1982, 53, 3813–3818.
5 S. Kahmann and M. A. Loi, J. Mater. Chem. C, 2019, 7, 2471–2486.
6 M. B. Price, J. Butkus, T. C. Jallicoe, A. Sadhanala, A. Briane, J. E. Dalpelt, K. Broch, J. M. Hodgkiss, R. H. Friend and F. Deschler, Nat. Commun., 2015, 6, 8420.
7 Y. Yang, D. P. Ostrowski, R. M. France, K. Zhu, J. van de Lagemaat, J. M. Luther and M. C. Beard, Nat. Photonics, 2016, 10, 53–59.
8 J. Yang, X. Wen, H. Xia, R. Sheng, Q. Ma, J. Kim, P. Tapping, T. Harada, T. W. Kee, F. Huang, Y.-B. Cheng, M. Green, A. Ho-Baillie, S. Huang, S. Shrestha, R. Patterson and G. Conibeer, Nat. Commun., 2017, 8, 14120.
9 P. Papagiorgis, L. Protesescu, M. V. Kovalenko, A. Othonos and G. Itskos, J. Phys. Chem. C, 2017, 121, 12434–12440.
10 M. Li, S. Bhaumik, T. W. Goh, M. S. Kumar, N. Yantara, M. Grätzl, S. Mhaisalkar, N. Mathews and T. C. Sum, Nat. Commun., 2017, 8, 14350.
11 J. Chen, M. E. Messing, K. Zheng and T. T. Pullerits, J. Am. Chem. Soc., 2019, 141, DOI: 10.1021/jacs.8b11867.
12 B. T. Diroll and R. D. Schaller, Adv. Funct. Mater., 2019, 29, 1901725.
13 J. Shah, Solid-State Electron., 1978, 21, 43–50.
14 R. F. Leheny, J. Shah, R. L. Fork, C. V. Shank and A. Migus, Solid State Commun., 1979, 31, 809–813.
15 J. M. Frost, L. D. Whalley and A. Walsh, ACS Energy Lett., 2017, 2, 2647–2652.
16 T. R. Hopper, A. Gorodetsky, J. M. Frost, C. Müller, R. Lovrinic and A. A. Bakulin, ACS Energy Lett., 2018, 3, 2199–2205.
17 J. M. Richter, F. Branchi, F. Valduga de Almeida Camargo, B. Zhao, R. H. Friend, G. Cerullo and F. Deschler, Nat. Commun., 2017, 8, 376.
18 T. Ghosh, S. Aharou, L. Etgar and S. Ruhman, J. Am. Chem. Soc., 2017, 139, 18262–18270.
19 T. R. Hopper, A. Gorodetsky, A. Jeong, F. Krieg, M. I. Bodnarchuk, M. Maimaris, M. Chaplain, T. J. Macdonald, X. Huang, R. Lovrinic, M. V. Kovalenko and A. A. Bakulin, Nano Lett., 2020, 20, 2271–2278.
20 R. A. Kerner, L. Zhao, Z. Xiao and B. P. Rand, J. Mater. Chem. A, 2016, 4, 8308–8315.
21 J. Endres, D. A. Egger, M. Kulbak, R. A. Kerner, L. Zhao, S. H. Silver, G. Hodes, B. P. Rand, D. Cahen, L. Kronik and A. Kahn, J. Phys. Chem. Lett., 2016, 7, 2722–2729.
22 F. Krieg, S. T. Ochenbein, S. Yakunin, S. ten Brinck, P. Aellen, A. Süess, B. Clerc, D. Guggisberg, O. Nazarenko, Y. Shynkarenko, S. Kumar, C.-J. Shihi, I. Infante and M. V. Kovalenko, ACS Energy Lett., 2018, 3, 641–646.
23 L. Protesescu, S. Yakunin, M. I. Bodnarchuk, F. Krieg, R. Caputo, C. H. Hendon, R. X. Yang, A. Walsh and M. V. Kovalenko, Nano Lett., 2015, 15, 3692–3696.
24 G. R. Yettapu, D. Talukdar, S. Sarkar, A. Swarnkar, A. Nag, P. Ghosh and P. Mandal, Nano Lett., 2016, 16, 4838–4848.
25 P. Guyot-Sionnest, M. Shim, C. Matranga and M. Hines, Phys. Rev. B: Condens. Matter Mater. Phys., 1999, 60, R2181–R2184.
26 F. T. Rabouw, R. Vaxenburg, A. A. Bakulin, R. J. A. van Dijk-Moes, H. J. Bakker, A. Rodina, E. Lifshitz, A. L. Efros, A. F. Koenderink, D. Vannmaelbergh, A. L. Efros, A. F. Koenderink, D. Vannmaelbergh, A. L. Efros, A. F. Koenderink and D. Vannmaelbergh, ACS Nano, 2015, 9, 10366–10376.
27 S. S. Lim, D. Giovanni, Q. Zhang, A. Solanki, N. F. Jamaludin, J. W. M. Lim, N. Mathews, S. Mhaisalkar, M. S. Pshenichnikov and T. C. Sum, Sci. Adv., 2019, 5, eaax3620.
28 K. T. Munson, C. Grieco, E. R. Kennehan, R. J. Stewart and J. B. Asbury, ACS Energy Lett., 2017, 2, 651–658.
29 J. Peng, Y. Chen, K. Zheng, T. Pullerits and Z. Liang, Chem. Soc. Rev., 2017, 46, 5714–5729.
30 W. P. D. Wong, J. Yin, B. Chaudhary, X. Y. Chin, D. Cortecchia, S.-Z. A. Lo, A. C. Grimsdale, O. F. Mohammed, G. Lancani and C. Soci, ACS Mater. Lett., 2020, 2, 20–27.
31 L. Protesescu, S. Yakunin, M. I. Bodnarchuk, F. Krieg, R. Caputo, C. H. Hendon, R. X. Yang, A. Walsh and M. V. Kovalenko, Nano Lett., 2015, 15, 3692–3696.
32 N. S. Makarov, S. Guo, O. Isaenko, W. Liu, I. Robel and V. I. Klimov, Nano Lett., 2016, 16, 2349–2362.
33 M. Wang and S. Lin, Adv. Funct. Mater., 2016, 26, 5297–5306.
34 L. D. Whalley, J. M. Skelton, J. M. Frost and A. Walsh, Phys. Rev. B, 2016, 94, 220301.
35 A. N. Beecher, O. E. Semonin, J. M. Skelton, J. M. Frost, M. W. Terban, H. Zhai, A. Alatas, J. S. Owen, A. Walsh and S. J. L. Billinge, ACS Energy Lett., 2016, 1, 880–887.
36 B. Li, Y. Kawakita, Y. Liu, M. Wang, M. Matsuura, K. Shibata, S. Ohira-Kawamura, T. Yamada, S. Lin, K. Nakajima and S. Liu, Nat. Commun., 2017, 8, 16086.
37 L. M. Herz, Annu. Rev. Phys. Chem., 2016, 67, 65–89.
38 M. Ortolani, A. Mancini, A. Budweg, D. Garoli, D. Breda and F. de Angelis, Phys. Rev. B, 2019, 99, 035435.
39 X.-Y. Zhu and V. Podzorov, J. Phys. Chem. Lett., 2015, 6, 4758–4761.
40 H. Zhu, K. Miyata, Y. Fu, J. Wang, P. P. Joshi, D. Niesner, K. W. Williams, S. Jin and X.-Y. Zhu, Science, 2016, 353, 1409–1413.
41 K. Miyata, D. Meggiolaro, M. T. Trinh, P. P. Joshi, E. Mosconi, S. C. Jones, F. De Angelis and X.-Y. Zhu, Sci. Adv., 2017, 3, e1701217.