Synthesis of Calcium Doped TiO2 Nanomaterials and Their Visible Light Degradation Property

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Abstract—Calcium with low charge density doped TiO2 (Ca-TiO2) mesoporous nanomaterials with high specific surface areas were successfully synthesized via a simple and effective solid-state reaction route. The properties of these nanomaterials were characterized by X-ray diffraction (XRD), small angle X-ray scattering (SAXS), scanning electron microscopy (SEM), energy dispersive spectroscopy (EDS), N2 absorption-desorption, ultraviolet visible light spectroscopy (UV-Vis) and infrared spectrum (IR). The results show that the samples possess the high specific surface area of 101.4m2/g, spherical nanosize of 9~16 nm, the structure of anatase TiO2, and the doped Ca2+ ion has been incorporated into the lattice or positioned on an interstitial lattice site of TiO2 with the changes of the amounts of calcium. Interestingly, the IR displays that the stretching vibrational peaks at ~1630cm−1 are assignable to the red shift of the -OH bond coming from the increasing of the doped calcium amount and the adsorption capacity of thiophene. Moreover, compared with the pure titanium dioxide, the Ca-TiO2 nanomaterials show excellent photocatalytic activities upon thiophene. For the Ca-TiO2 (Ca:Ti=1:7) photocatalyst, 90.2% of the initial thiophene can be degraded by visible light irradiation at 25°C for 50 min.

Keywords: Calcium doping; Titanium dioxide; Solid state reaction; Visible light degradation; Thiophene

I. INTRODUCTION

Thiophene is commonly regarded as a colorless, malodor, highly toxic, five membered heterocyclic compounds with a sulphur atom. It exists in soil and industry wastewater, which could migrate to groundwater, therefore, by bringing about a potential risk to human and animals. Although biological degradation of thiophene is found, the degradation rate is very slow [1], accompanied by the production of ill-defined detrimental compounds. Fortunately, recent studies reveal that semiconductor titania-mediated photodegradation of toxic compound is more attractive due to its low cost, strong oxidizing power and good chemical stability [2-4]. For instance, Yu et al. [5] found that 90% of pentachlorophenol could be degraded over the NiO/TiO2 composite. Liu [6] reported that the Congo red could be degraded by Al-doped TiO2 nanopowder irradiated by various lights. However, the large band gap (3.2eV) and relatively high recombination rate of photogenerated hole-electron pairs of anatase TiO2 have limited its widespread application in environmental protection. Up to date, effective approaches to overcome these challenges are to dope TiO2 with metal and nonmetal elements for enlarging the light absorption range of TiO2. Among these methods, transition metal, rare earth and noble metal ion doping [7-9] have been widely adopted to control the position of conduction band or valence band of TiO2, which could make the electrons excitable irradiated by visible light to generate the photoelectron-hole pairs. Besides, it is scarce to detaidedly investigate the alkaline-earth metal ion doping of TiO2 and their photocatalytic properties [10]. Noticeably, TiO2, as an n type semiconductor, when the electron density of the doped TiO2 decreases to some extents, could change into a p type semiconductor, which would change the photocatalytic activity [11]. To extend the absorption solar light range of TiO2 and to reduce the hole-electron recombination probability, calcium, in this work, was first employed to dop TiO2 for obtaining improved Ca-TiO2 mesoporous nanomaterials by a facile solid state reaction method in the broader sense and the effects of the calcium-doped content was investigated. Furthermore, the photodegradation efficiency of pure TiO2 and Ca-TiO2 nanopowder was comparatively studied by photodegradation of the thiophene solution.

II. EXPERIMENTAL

A. Materials

Tetrabutyl orthotitanate (TBOT, 98% (by mass)), Ca (NO3)2·4H2O (A.R.), thiophene (A.R.), cetyltrimethyl ammonium bromide (CTAB, A.R.) and pure TiO2 (surface area, 49.4m2/g) are provided by Shanghai Chemical Reagent Plant, China. All chemicals were used without
Further treatment.

B. Synthesis

The synthesis of Ca-TiO$_2$ photocatalyst is as follows: Ca(NO$_3$)$_2$·4H$_2$O and CTAB were firstly mixed and ground in a mortar for 15 min, then, the obtained mixture was mixed with TBOT and ground for 10 min. Then, the obtained mixture was mixed with TBOT and ground for 10 min. The feed molar ratio of Ca(NO$_3$)$_2$·4H$_2$O:TBOT is 1:4, 1:5, 1:6, 1:7, 1:8, 1:9, 1:10, 1:12, respectively, and CTAB is 10 wt% for the total weight of TBOT and Ca(NO$_3$)$_2$·4H$_2$O. The resulting viscous mixture was put into a beaker, followed by aging for 4 h and heated at 413 K in thermoelectric oven for 1.5 h to accelerate the reaction. After cooling to room temperature, the solid sample was ground and washed 5~8 times repeatedly with distilled water and then dried at 373 K. The samples were calcined in a muffle furnace at 823 K for 6 h with a heated temperature rate of 1 K. The samples were calcined in a muffle furnace at 823 K for 6 h with a heated temperature rate of 1 K.

C. Characterization

X-ray diffraction (XRD) studies of the sample were performed on a PANalytical diffractometer (CuK$_\alpha$ radiation, wavelength 0.15406 nm). The diffraction data were recorded for 20 angles between 5° and 70° with a scanning speed of 0.02°·s$^{-1}$. Scanning electron micrograph (SEM) and energy dispersive spectroscopy (EDS) were performed on a JSM-6700F electron microscope, with an accelerating voltage of 20 kV and a resolution of 1 nm. Nitrogen adsorption-desorption isotherm was obtained at 77 K with a Gemini-2390 VII type gas sorption and porosimetry instrument. Before the adsorption measurement, the sample was heated from room temperature to 413 K and evacuated to $10^{-3}$ Pa, and then kept at that temperature and vacuum outgas for 4 h. The specific surface area was determined by the BET method, assuming full surface coverage with nitrogen.

UV-Vis diffuse reflection spectra were measured using a UV-2550 spectrometer. The spectra were collected at 200-700 nm referenced to BaSO$_4$. The information of material surface structure property was carried out on the Perkin Elmer spectrum 65 infrared spectrometer with a spectral resolution of 1 cm$^{-1}$. The content of Ca was determined by the TAS-986 atomic absorption spectrophotometer with with analytical line $\lambda_{Ca}=422.7$ nm. The background was deducted by deuterium lamp; an air flow capacity of 6 L/min, a lamp electric current of 35 mA, a spectrum bandwidth of 0.1 nm, height of burner of 6 mm, and acetylene flow capacity of 2 L/min were used.

D. Photodegradation

The photocatalytic decomposition of thiophene over the photocatalysts was carried out in a XPA-VH photochemical reactor (Nanjing Xujiang Electromechanical Factory, China). A 300 W metal halogen lamp assisted with Toshiba B-47 optical filter ($\lambda>410$ nm) was used as a visible light source. In each run, the photocatalyst of 0.02 g was added into 20 mL of thiophene solution ($c_o=50$ mg/L) with pH of 7.45. Prior to the photodegradation experiment, the catalyst was allowed to reach a steady state with thiophene in dark for 60 min. Then, the adsorption capacity of the samples was determined. Later, the photodegradation reaction was initiated under visible light irradiation. The reaction system was kept continuously under magnetic stirring at a rate of 600 rpm, and the reaction temperature was controlled at 298 ± 1 K by a water-cooling system. The concentration of the residual thiophene ($c_t$) was measured at a fixed time interval by TU-1900 spectrometer at its maximum absorption wavelength of 229 nm.

III. Results and Discussion

A. XRD and SEM Analysis

The wide-angle XRD patterns of the samples are presented in Fig.1. The XRD patterns of all Ca-doped samples and pure TiO$_2$ can be assigned to anatase TiO$_2$ with reflection peaks in (101), (004), (200), (105), (211) and (204) crystal planes. The average crystallite sizes and crystal lattice strain of all the samples are given in Table 1, which are determined by the XRD software. From the data in table 1, it can be seen that all the crystallite sizes of Ca-TiO$_2$ samples are smaller than the pure TiO$_2$, indicating the doping of calcium effectively inhibits the growth of TiO$_2$ crystallites. Moreover, the average crystallite sizes ($L$) increase ($L_{Ca-TiO_2(1:4)} < L_{Ca-TiO_2(1:5)}$ ) and then gradually decrease in the following sequence of $L_{Ca-TiO_2(1:5)} > L_{Ca-TiO_2(1:6)} > L_{Ca-TiO_2(1:7)} > L_{Ca-TiO_2(1:8)} > L_{Ca-TiO_2(1:9)} > L_{Ca-TiO_2(1:10)} > L_{Ca-TiO_2(1:12)}$. Since the ion radius of Ca$^{2+}$ (0.099 nm) is larger than that of Ti$^{4+}$ (0.068 nm), the Ca$^{2+}$ ion doped enters into the lattice of TiO$_2$ with more difficulty than the other metal ion with similar ion radius of Ti$^{4+}$. In addition, when a Ca$^{2+}$ ion is located in a substitutional site of TiO$_2$, the electron density of the doped TiO$_2$ slightly decreases, and when decreasing of the feed molar ratio of Ca:Ti, the electron density of Ca doped TiO$_2$ sample relatively increases. Moreover, no other X-ray diffraction peaks are found in figure 1. The analytic results of the doped elemental component are listed in Table 1. For the alkaline-earth calcium metal, all above results confirm that

![XRD patterns for the Ca-TiO$_2$ samples](image-url)
the key approach is isomorphic substitution of the primary condition, and the limitant of ionic radius is relegated to secondary status in the heterovalent isomorphism circumstances. Typically, the scanning electronmicrograph image (the inset of fig.2 is EDS analysis of Ca-TiO$_2$ (1:4)) shown in fig.2 confirms that the sample Ca-TiO$_2$ (1:4) exhibits the shape of spherical particle and the particle sizes range change from 9 to 16 nm, which is consistent with the XRD analysis (10.7 nm).

**TABLE 1. THE CHARACTERISTIC PARAMETERS AND CONTENT OF THE SAMPLE**

| Sample       | Crystallite size /nm | Lattice stress /×10$^{-3}$ | BET /m$^2$/g | Bandgap /eV | Ca /at % |
|--------------|----------------------|---------------------------|--------------|-------------|----------|
| Ca-TiO$_2$(1:4) | 10.7                | 1.651                     | 101.4       | 2.986       | 3.67     |
| Ca-TiO$_2$(1:5) | 14.9                | 1.225                     | 89.6        | 3.048       | 3.12     |
| Ca-TiO$_2$(1:6) | 13.7                | 1.331                     | 92.3        | 3.077       | 2.62     |
| Ca-TiO$_2$(1:7) | 12.5                | 1.431                     | 89.3        | 3.200       | 2.05     |
| Ca-TiO$_2$(1:8) | 12.1                | 1.467                     | 93.5        | 3.127       | 1.75     |
| Ca-TiO$_2$(1:9) | 11.8                | 1.503                     | 78.1        | 3.144       | 1.23     |
| Ca-TiO$_2$(1:10) | 10.7                | 1.651                     | 73.2        | 3.190       | 0.92     |
| Ca-TiO$_2$(1:12) | 10.3                | 1.702                     | 69.8        | 3.193       | 0.81     |
| Pure TiO$_2$    | 25.4                | 0.068                     | 49.4        | 3.149       | 0        |

![Figure 2. SEM image of the sample Ca-TiO$_2$ (1:4).](image)

**B. UV-Vis Absorption Spectra**

Figure 3 illustrates the UV-vis absorption spectra of eight Ca-TiO$_2$ and pure TiO$_2$ samples. The small chart in the upper right corner of Fig.3 is an enlarged figure from 360 to 420 nm. Clearly for the absorption edge position of Ca-doped TiO$_2$ samples, a part of the samples move toward shorter wavelength, comparing with pure TiO$_2$ powders, another part of the samples exhibit red shift, meaning the band gap of Ca-doped TiO$_2$ samples decreases or increases after calcium doping. The band gap widening should be attributed to the decrease of TiO$_2$ crystal size [12]. From the table 1, when the molar ratio of Ca:Ti exceed 1:5, the size of TiO$_2$ crystal decreases from 14.9 to 10 nm, indicating that the band structure of TiO$_2$ in Ca-TiO$_2$ becomes quantized. Larger band gap energy of Ca-TiO$_2$ can result in larger thermodynamic driving force and accelerate charge carrier transfer rates in the normal Marcus region than its bulk phase counterparts. Moreover, the calcium doping can lead to the formation of partial Ti$^{3+}$, which limits the recombination rate of charge carriers. In addition, it can be observed from fig.3 that the absorbance of TiO$_2$ at 200-350nm increases with the decrease of doped calcium amount, and the visible light absorption range of Ca-doped TiO$_2$ samples also decreases correspondingly, which can result in the improvement of the photocatalytic activity in the UV region. The bandgap ($E_g$) can be calculated using the equation: $E_g$ (eV) = 1240 /λg, where λg is an intersection value for the vertical part and horizontal part of the spectrum. Fig. 4 shows the bandgaps of the samples. Obviously, the bandgap of Ca-doped TiO$_2$ samples increases with the decreases of molar ratio of Ca:Ti except for Ca:Ti=1:7, suggesting that the Ca-TiO$_2$ (1:7) photocatalyst has different performance for the photocatalytic activity.

![Figure 4. Bandgap of the Ca-TiO$_2$ samples.](image)

**C. IR Spectra**

Figure 5 shows the IR spectra of pure TiO$_2$ and Ca-doped TiO$_2$ samples. The broad peaks at ~3600, ~3100 and ~1630 cm$^{-1}$ correspond to the surface-adsorbed water and hydroxyl groups, respectively. As the molar ratio of Ca:Ti increases, the change of band shape is observed in two peaks in Fig.5. A distinct peak around 1040cm$^{-1}$ can be seen, which belongs to the Ca-O bond in the CaO crystal. However, this peak cannot be found on the undoped and low-content Ca-doped TiO$_2$ samples, which is agreement with the result of XRD. Another peak around 1390cm$^{-1}$ is present in the IR spectra of Ca-TiO$_2$ samples but absent for the pure TiO$_2$, and its intensity increases with the increase of calcium content. Therefore, the peak around 1140cm$^{-1}$ may be attributed to Ti-O-Ca bonds in this work. Interestingly, the intensity of vibration of hydroxyl displays difference each other, which indicates the existence of
show Ca element exists in the state of Ca$^{2+}$ after calcination.

### E. Adsorption and Photodegradation of Thiophene

Previous studies show that the anatase TiO$_2$ is found to be more active toward the photocatalytic degradation of organic substance than the rutile TiO$_2$ [15], so, we chose the anatase TiO$_2$ as the basis of our work. Considering the electronic absorption and surface area data (see table 1), thiophene concentration was monitored in the bulk solution for 60 minutes until adsorption/desorption equilibrium was reached, thus generating isotherms adsorption for the same concentration (50mg/L) of thiophene (see Fig.7). As shown in Fig.7, the maximum and minimal adsorption capacity of the Ca-TiO$_2$ (1:7) and pure TiO$_2$ samples were obtained, respectively. Another Ca-doped TiO$_2$ samples prepared in the same manner reveal that the thiphene adsorption capacity first increases and then decreases. When the molar ratio of Ca:Ti is greater than 1:7, thiphene adsorption capacity on the Ca-doped TiO$_2$ decreases, once the molar ratio of Ca:Ti is less than 1:7, thiphene adsorption capacity reduces. Based on the adsorption studies, the data from the visible light degradation of thiphene were obtained in Fig.7. For comparison, photocatalytic degradation of thiphene on commercial pure TiO$_2$ was also performed under the same conditions. Fig.7 shows that after irradiating under visible light for 50 min, 90.2% and 19.1% of thiphene degraded by the Ca-TiO$_2$ (1:7) photocatalyst and the pure TiO$_2$ are present, respectively. Moreover, the visible light degradation rate of thiphene is positive to adsorption capacity over the photocatalysts. Surprisingly and interestingly, the specific surface area of Ca-doped TiO$_2$ catalysts is less than two times of the pure TiO$_2$ (table 1), but the photocatalytic degradation rate of thiphene over Ca-doped TiO$_2$ photocatalysts is higher than twice of the pure TiO$_2$, indicating the surface area doesn’t exist in linear relationship with the degradation rates. Therefore, thiphene adsorption is not simply ascribed to the high surface area. Ulteriorly, the appropriate chemical compositions of Ca-doped TiO$_2$ mesoporous materials have played a significant role and thiophene is also sensitive to the calcium sites on these materials surface. Some researches indicate that the pollutant adsorption on the catalyst surface plays a very important role, and increasing the adsorption capacity of pollutant can improve the efficiency of photons during the photocatalytic process, leading to faster degradation of pollutant [16]. Thus, we conclude that degradation capacity of thiphene is directly relative to the distribution, quantity and vibration intensity of hydroxyl groups on the Ca-doped TiO$_2$ catalysts surface. Sulphur heteroatom with partial positive charge in thiphene combines easily with the hydroxyl groups, and its combination strength is proportional to quantity and distribution of hydroxyl groups. Evidently, these are in agreement with the results of infrared spectra. In addition, the Ca content in Ca-doped TiO$_2$ catalysts isn’t proportional to its photocatalytic activity, suggesting that a prerequisite for an effective dopant involves the possibility of charge trapping and migration to the surface of catalysts. The calcium ions doped are located mainly on the shallow...
surface of TiO$_2$ and can induce some defects. These defects could become the centers of shallow electrons or holes traps to improve efficiently the separation of the electrons and holes, resulting in higher photocatalytic activity of Ca-doped TiO$_2$ catalysts.

Figure 7. The adsorption and degradation of thiophene on the various Ca-TiO$_2$ samples.

IV. CONCLUSIONS

Ca-TiO$_2$ nanoparticles were successfully prepared by a simple solid-state reaction method. The results show that the Ca-TiO$_2$ sample presents an excellent visible light absorption and degradation property. The doped Ca contents in TiO$_2$ would be increased with augmenting of molar ratio of Ca:Ti. In addition, the doped Ca plays an essential role not only changing the visible absorption of TiO$_2$ but also decreasing particle size to increase the specific surface area. Ca-TiO$_2$ possess the outstanding degradation thiophene ability under visible light irradiation, being much superior to pure TiO$_2$, which was probably due to the enhanced thiophene absorption ability and effective depression of photogenerated hole-electron recombination.

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