Electrochemical Investigation of Adsorption of Single-Wall Carbon Nanotubes at a Liquid/Liquid Interface

Aminu K. Rabiu, Peter S. Toth, Andrew N. J. Rodgers, and Robert A. W. Dryfe

There is much interest in understanding the interfacial properties of carbon nanotubes, particularly at water/oil interfaces. Here, the adsorption of single-wall carbon nanotubes (SWCNTs) at the water/1,2-dichloroethane (DCE) interface, and the subsequent investigation of the influence of the adsorbed nanotube layer on interfacial ion transfer, is studied by using the voltammetric transfer of tetramethylammonium (TMA⁺) and hexafluorophosphat (PF₆⁻) as probe ions. The presence of the interfacial SWCNT layer significantly suppresses the transfer of both ions across the interface, with a greater degree of selectivity towards the PF₆⁻ ion. This effect was attributed both to the partial blocking of the interface by the SWCNTs and to the potential dependant adsorption of background electrolyte ions on the surface of the SWCNTs, as confirmed by X-ray photo-electron spectroscopy, which is caused by an electrostatic interaction between the interfacial SWCNTs and the transferring ion.

1. Introduction

Carbon nanotubes (CNTs) have attracted a great deal of attention owing to their interesting optical, mechanical, and electrical properties. They have potential application in a variety of areas such as nanoelectronics, field effect transistors, electrochemical and sensor devices, and as catalyst support. The modification of the CNT surface and as catalyst support for liquid/liquid interfaces, particularly the oil/water interface, have become increasingly popular for the assembly of a wide range of nanostructures, such as metal nanoparticles and two dimensional (2D) semiconducting nanomaterials.

The self-assembly of both single-wall carbon nanotubes (SWCNTs) and multi-walled carbon nanotube (MWCNTs) at a variety of liquid/liquid interfaces has also been explored greatly as an alternative method to generate functional CNT films. Typically, the material to be assembled at the interface (i.e. CNTs, in this case) is suspended in one of the bulk liquid phases. The suspension is then contacted with the second liquid phase, the liquid/liquid interface is referred to as the interface between two immiscible electrolyte solutions (ITIES). With only the oxidized form of the redox species [Ru(NH₃)₆]³⁺⁻ present in the aqueous phase, it was shown that, at the “bare” water/chloroform interface and attempted electrochemical characterization of the resultant interfacial layer by using scanning electrochemical microscopy (SECM). With the only oxidized form of the redox species [Ru(NH₃)₆]³⁺⁻ present in the aqueous phase, it was shown that, at the “bare” water/chloroform interface, a negative feedback current was generated as the tip approached the interface, owing to the interface acting as an insulator; whereas, in the presence of an Im-SWCNTs interfacial layer, a positive feedback current was generated at the tip, indicating that the Im-SWCNTs film was electroactive. However, as there was no redox species in the chloroform phase, no charge-transfer reaction occurred between the two immiscible liquids.

In the presence of appropriate electrolytes dissolved in each liquid phase, the liquid/liquid interface is referred to as the interface between two immiscible electrolyte solutions (ITIES). This special class of liquid/liquid interface can be polarized by the application of an external electric field, thus allowing both ion- and electron-transfer reactions to be readily studied by using electrochemical methods. The modification of the ITIES with adsorbed solids has been shown to be a viable means for studying the properties of interfacially adsorbed ma-

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terials, such as membrane porosity\cite{13} and catalytic activity of metal nanoparticles.\cite{14}

Recent studies in our laboratory have utilized this approach to probe the electrochemical properties of graphitic carbon nanostructures (CNTs and few-layer graphene) adsorbed at the ITIES. It was shown that interfacially assembled SWCNT/graphene layers serve as electron mediators, aiding heterogeneous electron transfer between aqueous and organic redox couples, which remain isolated in their respective phases.\cite{15a} This was utilized to functionalize interfacial SWCNT and graphene layers with metal nanoparticles by reducing aqueous metal salts using an organic electron donor,\cite{15a,16} and a conducting polymer polypyrrole,\cite{17} through oxidation of the pyrrole monomer dissolved in the organic phase by an aqueous oxidizing agent. Similarly, the electron-transfer-mediating properties of pristine liquid-phase exfoliated graphene at the water/organic interface were found to result in a catalytic effect on the heterogeneous oxygen reduction reaction.\cite{18} Furthermore, the electrochemical doping of the interfacial SWCNTs was investigated by using in situ Raman spectroelectrochemistry.\cite{15b}

The objective of the current work is to investigate the electrical properties of SWCNTs adsorbed at the water/DCE interface through analysis of their effect on the kinetics of ion transfer across the interface. The permeability of the films formed at different SWCNTs concentration by the ionic species is also described.

2. Results and Discussion

2.1. SWCNT Adsorption at Water/DCE Interface

Interfacial SWCNT layers were formed following a 10 min bath sonication of cells containing a DCE dispersion of SWCNTs and an aqueous phase solution. The SWCNT film located between the bulk phases was visible a few minutes after sonication. However, owing to some emulsification of both the water/DCE interface and the bulk liquid phases, caused by the sonication, the cells were left to stand for 12 h to allow the emulsion droplets in the bulk phases to coalesce before carrying out any electrochemical measurements. Figure 1a shows a typical cell 12 h after sonication.

The resultant interfacial SWCNT film was characterized in situ by using optical microscopy and ex situ by using scanning electron microscopy (SEM). Figure 1b shows an optical micrograph of a SWCNT film obtained by using a DCE dispersion concentration of 3 mg L\(^{-1}\). As can be seen, the interfacial film was composed of multiple emulsified droplets stabilized by SWCNTs. These droplets were observed to be stable for up to 7 days when left undisturbed. Longer time stability was not studied here. The droplets were also found to be stable when the water/DCE interface was polarized, as exemplified by the microscopy images in Figure 1c–e. The images showing the morphology of the interfacial layer at different applied potential differences across the water/DCE interface (\(\Delta\psi\)) were captured during a cyclic voltammetry experiment, where \(\Delta\psi\) was swept from \(-0.24\) to \(+0.46\) V. The only effect observed was the movement of the whole interfacial film towards one side

of the interface on positive polarization (indicated by the arrows, showing that as \(\Delta\psi\) was swept from \(-0.24\) to \(+0.46\) V, the space between the film and the wall of the glass increases) and vice-versa on reverse polarization. This movement of the SWCNT film may be connected with the movement of individual interfacial SWCNTs, as previously highlighted by the Girault group.\cite{5a} At extreme positive interfacial potential difference (\(\Delta\psi\) \(\geq\) +0.46 V), corresponding to background ion transfer, the interfacial film rotates clockwise on positive scan and anticklockwise on reverse (negative) scan. Figure 1e was taken during this rotation, which is the reason it appear like slightly out of focus.

Figure 2 shows ex situ SEM images of interfacial SWCNTs films prepared at different SWCNT dispersion concentrations. The interfacial films were carefully transferred onto a Si/SiO\(_2\) substrate prior to SEM measurement. As can be seen, the interfacial preparation method resulted in two types of SWCNT film morphologies depending on the initial dispersion concentration: at low SWCNT concentration (1 mg L\(^{-1}\)), the SWCNTs were predominately bent into rings (Figure 2a), with only a few straight or partially bent tubes, whereas at the higher SWCNT concentrations of 6, 12, and 18 mg L\(^{-1}\), porous interfacial films were formed, composed of random networks of multilayer SWCNTs (Figure 2b–d). The density of these multilayer films

![Figure 1.](image-url)
can be seen to increase with increasing nanotube concentration, although the density of the films obtained with 12 and 18 mg L$^{-1}$ CNT concentrations were very similar. The observed concentration-dependent transition of SWCNT morphologies from rings to straight tubes is similar to the findings of Wang et al. for a water/DCB Pickering emulsion system stabilized by SWCNTs.$^{[19]}$

2.2. Ion Transfer across SWCNT Interfacial Films

The composition of the cell employed for all electrochemical measurements is outlined in Scheme 1. Figure 3 shows the cyclic voltamograms (CVs) obtained in the presence of only the background electrolytes at the bare water/DCE interface and with interfacial SWCNT films prepared from two different bulk SWCNT dispersion concentrations (6 and 18 mg L$^{-1}$). It can be seen that, in the presence of the interfacial SWCNT layers, there was a slight increase in the capacitive current. This can be attributed to an increase in liquid/liquid interfacial roughness, owing to the presence of multiple emulsion droplets formed at the interface when the SWCNTs are adsorbed (Figure 1).$^{[20]}$ An alternative explanation is to consider the relative capacitances of the “free” and “blocked” parts of the interface; however, such an argument would lead to a decrease in the net capacitance, owing to the low capacitance of carbon nanotubes.$^{[21]}$ Further increases in the bulk SWCNT concentration resulted in very little increase in the capacitive current. Additionally, the interfacial SWCNT films affected the magnitude and shape of the background electrolyte ion transfer peaks (Li$^+$ and Cl$^-$), which limit the potential window on the positive and negative ends, respectively. The current magnitudes were reduced and the transfer peaks became broader, indicating that the presence of an interfacial SWCNT film makes the ion transfer more difficult.

2.3. TMA$^+$ and PF$_6^-$ Ion Transfer

The blocking effect of SWCNT interfacial films on ion transfer was further investigated by employing TMA$^+$ and PF$_6^-$ as probe ions. First, the transfer of each ion was performed in the absence of SWCNT layers and then repeated in the presence of interfacial SWCNT films of differing thickness. The CVs shown in Figure 4 were obtained for TMA$^+$ and PF$_6^-$ ions in the presence of SWCNT films prepared from bulk SWCNT concentrations of 1 mg L$^{-1}$. Also shown in the figure are CVs obtained at the unmodified interface for comparison. It can be seen that the responses of both TMA$^+$ and PF$_6^-$ ions were very similar to those obtained in the absence of the interfacial SWCNT films. There was only a small increase in peak separation ($\Delta E_p$) and a slight reduction in the peak current magnitudes.

However, when the nanotube dispersion concentration used for the film preparation was increased to 6 mg L$^{-1}$, the response obtained in all cases was significantly altered, as compared to those obtained at the bare interface (Figure 5); both forward and reverse transfer peaks were broadened and shifted away from each other and their magnitudes decreased. This behavior indicates that increasing the SWCNT dispersion concentration leads to a higher interfacial surface coverage, resulting in a greater part of the interface available for ion transfer being blocked by the SWCNTs.

Figure 6a shows a graph of the dependence of forward peak height ($I_p$), measured at bare and at SWCNT-covered interfaces, against the square root of the scan rate ($v^{1/2}$) for both TMA$^+$ and PF$_6^-$ ions. In each case, $I_p$ was linearly related to $v^{1/2}$ and the decrease in $I_p$ was similar for both ions. Using the Randles-Ševčík relation for the data collected from the bare water/DCE interface, the aqueous diffusion coefficient ($D_a$) of each ion was calculated. $D_a$ values obtained for TMA$^+$ (1.2×
Figure 4. CVs obtained for the transfer of TMA⁺ (a) and PF₆⁻ (b) in the absence (solid lines) and in the presence (dash lines) of SWCNT films prepared at a C_{SWCNT} of 1 mg L⁻¹. Scan rate was 50 mV s⁻¹.

Figure 5. CVs obtained for the transfer of TMA⁺ (a) and PF₆⁻ (b) in the absence (solid lines) and in the presence (dash lines) of SWCNT films prepared at a C_{SWCNT} of 6 mg L⁻¹. Scan rate was 50 mV s⁻¹.

Figure 6. a) Plot of forward transfer peak current for TMA⁺ and PF₆⁻ ions as a function of v^{1/2} in the absence (circles) and in the presence (squares) of a SWCNT film prepared by using a C_{SWCNT} of 6 mg L⁻¹. b) Change in ΔE_p (ΔE_p in the presence of SWCNTs – ΔE_p in the absence of SWCNTs) as a function of scan rate for each probe ion. C_{SWCNT} used for the assembly was 6 mg L⁻¹.

10⁶ cm² s⁻¹) and PF₆⁻ (1.4 × 10⁶ cm² s⁻¹) were in agreement with the literature values of 1.2 × 10⁶ cm² s⁻¹ [22] for TMA⁺ and 1.5 × 10⁶ cm² s⁻¹ [23] for PF₆⁻. Figure 6b shows the change in ΔE_p for each ion as a function of scan rate in the presence of SWCNTs.

It can be seen that the change in ΔE_p is greater for PF₆⁻ compared to TMA⁺, which indicates that the kinetics of the PF₆⁻ transfer were more inhibited by the interfacial SWCNT film.

To rationalize this observed ion selectivity, the possible adsorption of either the probe ions or the organic background electrolyte ions on the assembled SWCNT film was investigated by using chronoamperometry and XPS. Firstly, potential step experiments were performed for each probe ion transfer. The interfacial potential was stepped from a potential where no ion transfer occurs (−0.1 and +0.25 V for TMA⁺ and PF₆⁻, respectively) to a potential where ion transfer from the water to organic phase takes place (+0.25 and −0.18 V for TMA⁺ and PF₆⁻, respectively). The interfacial potential was held at the ion transfer potential for 10 min, after which the SWCNT film was carefully transferred onto a Si/SiO₂ wafer. The transferred layers were then washed in ethanol, isopropanol, and acetone and dried before subsequent XPS analysis. A control sample was treated in a similar way, with the exception that neither the probe ions nor the supporting electrolyte ions were present and no interfacial potential was applied.

Figure 7a presents XPS spectra of the aforementioned SWCNT films. The spectra show the presence of B, N, Cl, and P in films obtained with either TMA⁺ or PF₆⁻ present, but not in the control sample. The appearance of signals attributable to B, Cl, and P in the TMA⁺ and PF₆⁻ samples is indicative of ad-
The apparent rate constant \( k_{\text{app}} \) of TMA\(^{+}\) and PF\(_6^-\) ion transfer in the presence of SWCNTS films was determined by using the Nicholson method\(^{[25]}\). \( \Delta E_p \) values measured at scan rates higher than 25 mVs\(^{-1}\) were used. Figure 8 shows the \( k_{\text{app}} \) values obtained. Increasing the SWCNT dispersion concentration in the organic phase resulted in a decrease in \( k_{\text{app}} \) for the TMA\(^{+}\) ion, owing to the greater surface coverage by the SWCNTS (Figure 2). This can be explained by invoking Amatore’s theory of voltammetry\(^{[26]}\) at a partially blocked electrode if we assume that the SWCNTS have transformed the single continuous interfacial area into a large number of smaller randomly distributed micro-/nanopores, the size and/or density of which decreases with increased interfacial coverage. According
to the theory, under conditions of total overlap of the diffusion layers, \( k_{\text{app}} \) is lowered by a factor of \((1 - \theta)\) [Eq. (1)]:

\[
k_{\text{app}} = k_0 (1 - \theta)
\]

where \( \theta \) is defined as the fractional area covered by the blocking nanotube film.

The fact that the voltammetric profile of TMA\(^+\) transfer exhibited a peak-shaped response rather than a sigmoidal one indicates that an overlapping linear diffusion field was achieved. The interfacial coverage was estimated from the SEM data to be about 77.5, 88.0, and 91.7% when the SWCNT concentration used in film preparation was 6, 12, and 18 mg L\(^{-1}\) respectively. Therefore, applying the \((1 - \theta)\) correction factor gave an average \( k_0 \) value of 1.0 \( \pm 0.1 \times 10^{-2} \) cm s\(^{-1}\).

Aside from the slower kinetics displayed by the PF\(_6^-\) ion compared to TMA\(^+\) transfer in the presence of interfacially assembled SWCNTs (Figures 6b and 8), it is also clear from Figure 8 that the negative probe ion also shows a less clear dependence of \( k_{\text{app}} \) on SWCNT concentration. The ion transfer and the XPS data indicate that there is a potential-dependent change in surface composition of the nanotubes, which in turn suggests that the nanotubes adsorb on the interface from the organic phase, that is, they constitute part of organic double layer. This effect is then associated with the high surface charge density exhibited by the interfacial SWCNTs at the PF\(_6^-\) transfer potential, which leads to the attainment of maximum blockage at the SWCNT concentration of 6 mg L\(^{-1}\), as against the TMA\(^+\) ion.

3. Conclusions

In the present study, we have demonstrated the use of ion transfer voltammetry at the liquid/liquid interface to characterize the electrical properties of SWCNTs adsorbed at a water/DCE interface. In the presence of adsorbed SWCNT layers of varying density/thickness, transfer of the positively charged TMA\(^+\) ion across the interface was found to be less inhibited than the corresponding negative PF\(_6^-\) ion. The retardation of ion transfer by the nanotube layer was analyzed using the theory of voltammetry at partially blocked electrodes, and the selectivity between TMA\(^+\) and PF\(_6^-\) ions was attributed to the potential-dependent adsorption of the organic supporting electrolyte ions on the interfacial SWCNTs, as indicated by XPS measurements, which caused electrostatic interaction between the transferring ion and the SWCNT surface and, thereby, inhibited the ion transfer.

Experimental Section

Materials

Arc discharge SWCNTs (purified, > 95% carbon), 1,2-dichloroethane (DCE, ≥ 99.8%), lithium chloride (LiCl, ≥ 99%), tetramethylammonium chloride (TMACl, ≥ 99%), sodium hexafluorophosphate (NaPF\(_6\), 99.99%), potassium chloride (KCl, 99.8%) potassium tetrakis(4-chlorophenyl) borate (KTPBACl, ≥ 98%), and bis(triphenylphosphoranylidene) ammoniumchloride (BTPPACl, 97%) were purchased from Sigma Aldrich and used as received. Dibenzo-18-crown-6 (98% +) was a product of Lancaster Synthesis. The bis(triphenylphosphoranylidene) ammonium tetrakis(4-chlorophenyl) borate (BTPPATPBCl) used as the organic-phase electrolyte was prepared as described previously.[18,27] Ultrapure water (18.2 MΩ cm resistivity, Milli-Q Direct 8, Merck Millipore, USA) was used for aqueous solutions preparation.

Methods

SWCNT dispersions in DCE were prepared by sonication. Pristine SWCNTs (22 mg) were placed in a 500 mL flat-bottom glass bottle containing DCE (100 mL). The contents were bath sonicated for 24 h by using an Elmasonic P70 H sonicator (Elma GmbH & Co. KG) at 37 KHz and 30% power setting. The as-prepared dispersion was stable for months. Aliquots of the dispersion were taken, diluted, and used to determine the extinction coefficient (\(a\)) by using UV/Vis absorption spectroscopy. The value of \(a\) obtained at 660 nm was 39 \( \pm 0.9 \times 10^3\) mg \( \cdot\) mL\(^{-1}\) and agrees with 41.00 \( \pm 0.4 \times 10^3\) mg \( \cdot\) mL\(^{-1}\) reported previously for CVD-grown SWCNTs is reported previously for CVD-grown SWCNTs[16] and was similar to that reported elsewhere.[28] Optical images of SWCNT interfacial films were recorded with a stereozoom microscope (SMZ168, Motic) connected to a digital live camera (GXCAM-9, GX Optical). SEM images were obtained by using an FEI XL30 Environmental SEM–FEG operated under high-vacuum state with an accelerating voltage of 15 keV. XPS was performed by using a K-Alpha X-ray photoelectron spectrometer (Fisher scientific) located at the EPSRC NEXUS facility, Newcastle University, UK. The survey spectra were taken at 0.4 eV step size at three different locations on each sample. All experiments were carried out at room temperature.

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Conflict of interest

The authors declare no conflict of interest.
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