Improved Anti-Biofouling Performance of Thin-Film Composite Forward-Osmosis Membranes Containing Passive and Active Moieties

Longbin Qi,†‡§ Yunxia Hu,*∥ Zhongyun Liu,‡ Xiaochan An,‡§ and Edo Bar-Zeev‖

†State Key Laboratory of Separation Membranes and Membrane Processes, School of Materials Science and Engineering, Tianjin Polytechnic University, Tianjin 300387, PR China
‡CAS Key Laboratory of Coastal Environmental Processes and Ecological Remediation; Research Center for Coastal Environmental Engineering and Technology of Shandong Province; Yantai Institute of Coastal Zone Research, Chinese Academy of Sciences, Yantai, Shandong Province 264003, PR China
§University of Chinese Academy of Sciences, Beijing 100049, PR China
‖Department of Environmental Hydrology & Microbiology, Zuckerberg Institute for Water Research (ZIWR), Ben-Gurion University of the Negev, Beersheba, 8499000 Israel

Supporting Information

ABSTRACT: Forward osmosis (FO) has gained increasing attention in desalination, wastewater treatment, and power generation. However, biofouling remains a major obstacle for the sustainable development of the FO process. Both passive and active strategies have been developed to mitigate membrane biofouling. A comprehensive understanding of different strategies and mechanisms has fundamental significance for the antifouling membrane development. In this study, thin-film composite (TFC) FO membranes were modified with polydopamine (PDA) coating as a passive antibacterial moiety and silver nanoparticles (Ag NPs) as an active antibacterial moiety. Their anti-biofouling performances were investigated both in static and dynamic conditions. In static exposure, the PDA-coated membranes exhibited great passive anti-adhesive property, and the Ag-NP-generated membranes presented both of excellent passive anti-adhesive properties and active antibacterial performance. While in dynamic cross-flow running conditions, Ag NPs effectively mitigated the membrane water flux decline due to their inhibition of biofilm growth, the PDA coating failed because of its inability to inactivate the attached bacteria growth. Moreover, Ag NPs were stable and active on membrane surfaces after 24 h of cross-flow operation. These findings provide new insights into the performances and mechanisms of passive and active moieties in the FO process.

INTRODUCTION

Increasing demand on freshwater and clean energy is one of the greatest global challenges due to the rapidly population growth, growing urbanization, and changing global climate.1−2 The forward-osmosis (FO) membrane process works as a state-of-the-art technology to desalinate seawater,3−4 reuse wastewater,5−7 and produce clean energy using salinity gradients,8−9 alleviating freshwater shortages and clean energy crises locally and globally. The development of high-performance thin-film composite (TFC) FO membranes has gained increasing attention in the pursuit of achieving high water flux without compromising salt rejection during long-term operation under environmental conditions.10−12 However, membrane fouling (and particularly biofouling) remains a bottleneck for the development and application of large-scale FO systems, which significantly reduces membrane performance and increases operation costs as well as energy inputs.13−15

Passive and active strategies have been developed to mitigate biofilm by minimizing the deposition and growth of microorganisms on membrane surface.16−18 The passive strategy generally involves membrane surface modification by grafting19−21 coating, or blending hydrophilic materials22,23 to weaken the foulant–membrane interactions and prevent foulants adsorption on membrane surface.16,24 The active strategy normally incorporates biocides on membrane surface to deactivate bacteria and prevent biofilm formation.25−27 Foulant adsorption on membrane surface is an initial stage of
membrane fouling. The passive strategy presents effective performances against fouling under low or mild fouling conditions. However, under severe fouling conditions, the passive strategy may lose its efficacy because the membrane surface is quickly masked by the foulant layer, and the foulant–foulant interaction is dominant to cause membrane fouling. Moreover, biofouling is more severe and complicated than inorganic and organic fouling because microorganisms are living foulants to secrete sticky proteins and polymers and induce strong interaction with membrane surfaces. The membrane performances may not be impaired if the attached microorganisms can be deactivated by biocides and then washed off easily. Therefore, an ideal anti-biofouling FO membrane should have both passive and active functions to minimize bacterial absorption and deactivate bacterial growth on the membrane surface. A holistic understanding of different strategies and mechanisms has fundamental significance for the antifouling membrane design.

However, few strategies have been developed to incorporate both passive and active moieties on membrane surfaces for the fabrication of ideal anti-biofouling FO membranes. Our previous work and other reported research have demonstrated that mussel-inspired dopamine chemistry provides a simple yet universal approach to incorporating both passive and active moieties on most types of inorganic and organic substrates. Dopamine can self-polymerize into polydopamine (PDA) on various substrates with controllable film thickness and durable stability. PDA has been confirmed to exhibit antifouling properties of alleviating the absorption of protein and microorganisms. Silver nanoparticles (Ag NPs) can form in situ on the PDA coating upon incubation with silver salt solution and act as an efficient biocide. Thus, dopamine chemistry provides a facile process by which to incorporate both passive PDA coating and active Ag NPs on FO membrane surfaces for antibiofouling properties. However, no work has been done to investigate and compare the passive and active anti-biofouling performances of PDA and Ag-NP-functionalized TFC FO membranes in the continuous cross-flow forward-osmosis test system.

Herein, the thin-film composite (TFC) FO membranes were fabricated and then modified with PDA coating as a passive antibacterial moiety and Ag NPs as an active antibacterial moiety. Their anti-biofouling performances were investigated in both static and dynamic conditions and also compared on both operation modes of FO (active-layer-facing feed solution) and pressure-retarded osmosis (PRO, active layer facing draw solution) to gain comprehensive an understanding of the anti-biofouling performances induced by PDA and Ag NPs. Pseudomonas aeruginosa was selected as a model bacterial stain, and synthetic wastewater was used as a low osmotic feed solution. Dynamic biofouling experiments were conducted in a continuous cross-flow FO system. The structure and composition of biofilms were analyzed to clarify the mechanisms of biofouling mitigation by PDA and Ag NPs. Our findings aim to provide new insights on the performances and mechanisms of passive and active moieties on the anti-biofouling efficiency of TFC FO membranes.

**MATERIALS AND METHODS**

**Materials and Chemicals.** A commercial polyester nonwoven fabric (PET; grade 3249, 40 μm thickness) was purchased from Ahlstrom (Helsinki, Finland). Polysulfone (PSf, Mn: 22 000 Da), m-phenylenediamine (MPD, >99%), 1,3,5-benzenetricarbonyl trichloride (TMC, 98%), N-methyl pyrrolidone (NMP), and dopamine were supplied by Sigma-Aldrich (Saint Louis, MO). SYTO9 green fluorescent nucleic acid stain and Concanavalin A (Con A, Alexa Flour 633) were obtained from Invitrogen (Eugene, OR), and propidium iodide (PI) was received from Sigma-Aldrich. Tris–HCl buffer (1 M, pH 8.5) was purchased from Beijing Solarbio Science & Technology Co., Ltd. P. aeruginosa (ATCC 27853) was obtained from the American Type Culture Collection. Yeast extract, tryptone, and agar were purchased from Oxoid Limited. AgNO₃, NaCl, MgSO₄·7H₂O, NaHCO₃, CaCl₂·H₂O, KH₂PO₄, NH₄Cl, EDTA, and sodium citrate were purchased from Sinopharm Chemical Reagent Beijing Co., Ltd., China. All chemicals were used as received.

**TFC Membrane Fabrication.** TFC FO membranes were fabricated following our previous protocol through two steps including the casting of a PSf ultra-filtration support layer by a typical non-solvent-induced phase separation (NIPS) method and the formation of polyamide active layer by interfacial polymerization on top of PSf support. PSf casting solution (12 wt %) in NMP was cast on a PET-integrated PSf support layer. A wet PSf membrane was taped to a clean glass plate covered by PET nonwoven fabric and precipitated in a water coagulation bath at room temperature to produce a PET-integrated PSf support membrane. The casting process, the humidity was controlled at 40 ± 5 RH% in a clean room. Polyamide layer was formed by interfacial polymerization of MPD in the aqueous phase and TMC in the organic phase on top of the PSf support layer. A wet PSf membrane was tapped to a clean glass plate and then immersed in 30 mL of aqueous solution with 3.4 wt % MPD for 2 min. A rubber roller was used to remove excess MPD solution from the PSf membrane surface. The MPD-containing PSf membrane was then immersed in a 0.15 wt % TMC solution in hexane for 1 min to form polyamide films. Finally, the membranes were cured in DI water at 95 °C for 2 min. After thorough rinsing with DI water, the TFC FO membranes were stored in DI at 4 °C before characterization.

**TFC Membrane Surface Functionalization.** In situ growth of Ag NPs on both surfaces of TFC FO membranes was performed according to our previous work. Briefly, the TFC FO membrane coupons were immersed in a 2 mg/mL dopamine solution at 10 mM Tris–HCl buffer (pH 8.5) for 1 h at room temperature under 60 rpm shaking to fabricate PDA-coated membranes. To generate Ag NPs on membrane surfaces, the PDA-coated membrane coupons were immersed in 50 mM silver nitrate aqueous solution at room temperature, and the incubation time was varied (1, 2, 4, or 8 h) to optimize the generation conditions of Ag NPs. Finally, the membrane coupons were rinsed three times with DI water, air-dried at room temperature, and stored in a light-proof dark chamber for further characterization.

**TFC Membrane Characterization.** The surface morphologies of pristine and modified membranes were observed using a scanning electron microscope (SEM, S-4800, Hitachi). Ag NPs on the membrane surfaces were detected using an EX-350 energy-dispersive X-ray microanalyzer (EDX, Horiba, Tokyo, Japan).
wastewater composed of 8.0 mM NaCl, 0.15 mM MgSO₄ 7.5) was prepared based on the reported protocol. Then incubated with exponential growth phase. Bacteria were collected after 1 min with sterile synthetic wastewater to remove unbound stains. Live and dead cells were stained with 3.34 μM SYTO9 and 20 μM PI solution, respectively, for 30 min in dark to label live cells, dead cells, and silver ions in the solutions were measured completely dissolve Ag NPs in the membrane. The total organic carbon (TOC) measurement was performed using ICP-MS (ELAN DRC II, PerkinElmer). Membrane coupons were immersed in 10 mL of 3.5% HNO₃ solution for 48 h to completely dissolve Ag NPs in the membrane. The concentrations of silver ions in the solutions were measured using ICP-MS (ELAN DRC II, PerkinElmer).

Static Anti-Adhesion Tests. P. aeruginosa was used as a model bacterial strain to evaluate the bacteria adhesion and viability on both surfaces of the TFC FO membrane, following our previous protocol with slight changes. P. aeruginosa was initially cultured in Luria–Bertani (LB) broth to reach a mid-exponential growth phase. Bacteria were collected after 1 min of centrifugation at 5000 rpm and then resuspended in sterile synthetic wastewater solution to reach an optical cell density at 600 nm (OD₆₀₀) of 0.15, where the initial concentration of P. aeruginosa was approximately 10⁶ CFU mL⁻¹. Synthetic wastewater composed of 8.0 mM NaCl, 0.15 mM MgSO₄ 7H₂O, 0.5 mM NaHCO₃, 0.2 mM CaCl₂·H₂O, 0.2 mM KH₂PO₄, 0.4 mM NH₄Cl, and 0.6 mM sodium citrate (pH 7.5) was prepared based on the reported protocol.

Bacterial attachment was evaluated by placing circular membrane coupons with the diameter of 1.6 cm were placed in sterile plastic tubes with 5 mL of P. aeruginosa suspension at 37 °C for 1 h. Biofouled membrane coupons were then collected and gently rinsed using sterile synthetic wastewater to remove unattached cells. Live and dead cells were stained with 3.34 μM SYTO9 and 20 μM PI solution, respectively, for 30 min in the dark. The membranes were then rinsed three times with sterile synthetic wastewater to remove unbound stains. The fluorescence images of the stained membrane coupons were collected using a confocal laser scanning microscopy (CLSM, Fluo View FV1000, Olympus). SYTO9 was excited with an argon laser at 488 nm, and PI was excited with a diode-pumped solid-state laser at 559 nm. Live cells and dead cells were recognized as green and red spots, respectively. The CLSM images were analyzed using ImageJ Pro software (National Institutes of Health). For quantitative image analysis, three individual samples were measured in a parallel manner, and the results were reported as the average values with the standard deviations.

Antimicrobial Activity Measurements. The antimicrobial activity of Ag NPs-containing TFC FO membranes was assessed using the reported colony-forming unit (CFU) method. Circular membrane coupons with 1.6 cm in diameter were cut from the freshly prepared membranes and then incubated with P. aeruginosa suspensions (10⁸ CFU mL⁻¹) in the mid-exponential growth phase for 5 h at 37 °C. After gentle rinsing with sterile physiological saline solution (0.15 M NaCl, 20 mM NaHCO₃, pH 7.5), the bacteria attached on the membrane coupons were removed through 7 min bath-sonication in 10 mL of sterile physiological saline solution. The bacterial suspensions were serially diluted 100 times, and 100 μL of the dilution was taken to spread onto an LB agar plate. Then bacteria colonies were counted after 24 h of incubation at 37 °C. The antibacterial efficiency \(E_b\) was calculated by the following equation:

\[
E_b = \left( \frac{N_p - N_m}{N_p} \right) \times 100\% 
\]

where \(N_p\) and \(N_m\) are the numbers of colonies formed on pristine membranes and modified membranes, respectively.

Dynamic Biofouling Experiments and Biofilm Characterization. The dynamic biofouling measurements were carried out in a lab-scale FO cross-flow setup with the effective surface area of 8 cm² reported in our previous work. The FO setup was stabilized using DI water as both feed solution and draw solution to obtain the water flux of 0 for 20 min. The synthetic wastewater stock and NaCl stock was added into the feed solution side and draw solution side separately to achieve a permeate water flux of 20 L m⁻² h⁻¹. P. aeruginosa suspension was then added into the feed solution side to reach a bacterial concentration of 6.0 × 10⁷ CFU L⁻¹. The volumes of feed and draw solutions were 2 L for all of the experiments. The pH of feed solution was adjusted to pH 7.5 during the dynamic biofouling experiment. The cross-flow rates of both feed and draw solutions were maintained at 0.5 L min⁻¹ and the temperatures of solutions were kept at 25 ± 0.5 °C. The water flux was continuously monitored by measuring the weight increment of the draw solution using an electronic balance (ME3002, Mettler, Switzerland). The FO filtration tests were operated under both FO and PRO modes, respectively.

Biofilms were analyzed upon staining using CLSM. Briefly, 1 cm² of biofouled membrane coupon was cut from the tested membranes and then stained with SYTO9, PI, and 50 μM Con A for 30 min in dark to label live cells, dead cells, and extracellular polymeric substance (EPS), respectively. A Z-stack scanning with a slice thickness of 2 μm was performed to collect the images of biofilm. The three-dimensional structure of biofilm was reconstructed using the FV1000 Viewer software (Olympus). Con A was excited with a diode-pumped solid-state laser at 633 nm.

Stability of Ag NPs on TFC Membrane. To evaluate the stability of Ag NPs on TFC membrane, the membranes were analyzed using CLSM. Briefly, 1 cm² of biofouled membrane coupon was cut from the tested membranes and then stained with SYTO9, PI, and 50 μM Con A for 30 min in dark to label live cells, dead cells, and extracellular polymeric substance (EPS), respectively. A Z-stack scanning with a slice thickness of 2 μm was performed to collect the images of biofilm. The three-dimensional structure of biofilm was reconstructed using the FV1000 Viewer software (Olympus). Con A was excited with a diode-pumped solid-state laser at 633 nm.

Total Organic Carbon and Total Protein Analysis. The total organic carbon (TOC) measurement was performed according to the reported protocol, and membrane coupons were cut (1 cm × 1 cm) from the center of biofouled membranes and placed into sterile glass bottles with 20 mL of
hydrogen chloride aqueous solution (20 mM). The biofilm was removed from the membrane coupons through 60 s of sonication using a probe sonicator (JY92-IIDN, Scientz biotechnology Ning Bo Co., Ltd., China), and TOC in the solution was quantified using a TOC analyzer (TOC-VCPH, Shimadzu, Japan) according to the operation manual of manufacture. The TOC of biofilm was reported as the average value from three parallel samples and the standard deviation as the error bar after normalization by the surface area of membrane coupons.

Total protein biomass was quantified using the reported BCA method.44,45 Biofouled membrane coupons (1 cm × 1 cm) were placed into 2 mL Eppendorf tubes with 1 mL of PBS buffer and then sonicated for 7 min to break bacteria cells. After centrifugation at 13 000 rpm for 10 min, the protein in the supernatant solution was quantified using a BCA protein assay kit (Institute of Biological Engineering Co., Ltd., China) according to the operation manual of manufacture. The total protein biomass of biofilm was reported as the average value from three parallel samples and the standard deviation as the error bar after normalization by the surface area of membrane coupons.

RESULTS AND DISCUSSION

PDA Coating and Ag NPs Generation and Increase of Surface Hydrophilicity without Affecting the Transport Properties of TFC FO Membranes. Our previous work had demonstrated that Ag NPs were in situ generated on both surfaces of TFC FO membranes, including the top polyamide surface and the bottom PET-strengthened polysulfone surface via mussel-inspired dopamine chemistry.33 Here, we further investigated the optimized experimental conditions of in situ Ag NPs generation on both surfaces of TFC FO membranes. SEM analysis indicates that the PDA coating did not present morphology change between the PDA coated membranes (Figure 1b,e) and pristine membranes (Figure 1a,d) on the polyamide surface and the polysulfone surface. Ag NPs were densely populated and uniformly distributed on characteristic ridge-and-valley polyamide surface (Figures 1c and S1) as well as a smooth polysulfone surface (Figure 1f). Interestingly, the Ag NPs generated on the polysulfone surface was 28.1 ± 4.7 nm in diameter, which is bigger than 10.6 ± 2.5 nm in diameter of Ag NPs formed on the polyamide surface. This change in Ag NPs size is likely due to the different surface morphologies and properties of polysulfone surface and polyamide surface, which affect the growth of PDA coating and thus tailor the nucleation and growth of Ag NPs.33,46 In addition, the Ag NPs generated on the polysulfone surface presented stronger silver signals than that on the polyamide surface, indicating the greater silver mass deposited on the polysulfone surface.

Ag NPs generation on both surfaces of TFC FO membranes. SEM analysis indicates that the PDA coating did not present morphology change between the PDA coated membranes (Figure 1b,e) and pristine membranes (Figure 1a,d) on the polyamide surface and the polysulfone surface. Ag NPs were densely populated and uniformly distributed on characteristic ridge-and-valley polyamide surface (Figures 1c and S1) as well as a smooth polysulfone surface (Figure 1f). Interestingly, the Ag NPs generated on the polysulfone surface was 28.1 ± 4.7 nm in diameter, which is bigger than 10.6 ± 2.5 nm in diameter of Ag NPs formed on the polyamide surface. This change in Ag NPs size is likely due to the different surface morphologies and properties of polysulfone surface and polyamide surface, which affect the growth of PDA coating and thus tailor the nucleation and growth of Ag NPs.33,46 In addition, the Ag NPs generated on the polysulfone surface presented stronger silver signals than that on the polyamide surface, indicating the greater silver mass deposited on the polysulfone surface.

PDA imparts its hydrophilicity to membrane surfaces once its coating forms because it decreased the water-contact angles of both the polyamide layer and PSf support from 70.8 ± 4.6 and 125.4 ± 3.6° to 43.8 ± 5.7 and 79.1 ± 8.7°, respectively, as shown in Table S1. The Ag NP generation further decreased the water contact angles of polyamide layer and PSf support to 37.9 ± 3.8° and 75.1 ± 8.6°, respectively, and, thus, improved the membrane surface hydrophilicity because of the hydrophilic nature of Ag NPs and the improved surface roughness of Ag NPs modified membranes.47,48 Noticeably, the pristine PSf support surface showed a quite-high water-contact angle due to the hydrophobic PET nonwoven fabric and polysulfone on the back surface of the TFC FO membrane (Figure S2).
Furthermore, the ζ-potential measurements shown in Table S1 found that the PDA coating and Ag NPs generation did not affect the ζ potentials of TFC FO membranes and, therefore, did not change their surface charges. The impacts of PDA coating and Ag NPs on the transport properties of TFC FO membrane were investigated to optimize surface modification conditions. PDA coating had no major effect on membrane separation performance within 1 h of modification due to the formation of thin PDA layer. Under the condition of 1 h of PDA coating, Ag NP generation had a negligible effect on membrane water fluxes within 2 h of modification and then resulted in 10% flux reduction with increasing the incubation time to 8 h (Figure S3a). In addition, the reverse salt flux of the TFC FO membrane was not affected by Ag NPs generation. Under the optimized conditions of 1 h of PDA coating and 2 h of AgNO₃ immersion time, the PDA-coated and Ag-NP-generated TFC FO membranes were prepared, and their intrinsic transport parameters were measured following the reported protocol. The results shown in Figure S3b illustrate that the PDA coating and Ag NPs generation had no significant impact on membrane transport properties.

**PDA Coating and Ag NPs Imparting of Membrane Anti-Adhesive and Antimicrobial Properties.** Bacteria attachment on membrane surface is a critical step of biofilm formation and development during biofouling. Thus, the investigation of bacterial adhesion on membrane surface is very conducive to understanding the mechanism of biofouling control. As shown in Figure 2a, the total bacterial cells attached on the membrane surface decreased by 70% from 7.6 × 10⁵ cells per square centimeter to 2.3 × 10⁵ cells per square centimeter for the pristine and PDA-coated polysulfone surfaces of TFC FO membranes and (b) total number and (c) viability of P. aeruginosa cell attached on membrane surfaces. Results are presented as an average of at least three different membrane samples, and error bars are given as standard deviations. Bacterial cells were stained with SYTO9 (green) and PI (red) for live and dead cells, respectively. Scale bars in CLSM are 10 μm. Asterisks indicate statistical significance determined by a Student t test (p value of <0.05).

![Figure 2](image.png)

**Figure 2.** Static anti-adhesion properties of TFC FO membranes. (a) Representative CLSM images of P. aeruginosa cells attached on the pristine, PDA-coated, and Ag-NP-generated polysulfone surfaces of TFC FO membranes and (b) total number and (c) viability of P. aeruginosa cell attached on membrane surfaces. Results are presented as an average of at least three different membrane samples, and error bars are given as standard deviations. Bacterial cells were stained with SYTO9 (green) and PI (red) for live and dead cells, respectively. Scale bars in CLSM are 10 μm. Asterisks indicate statistical significance determined by a Student t test (p value of <0.05).

Even though the surface charge of membrane surface is negative at pH 7.5, the protonation degree of amine groups in PDA was reported as 94%. The smaller viability of bacteria on PDA-coated membranes is not only due to the antibacterial properties of PDA but also due to the less number of bacteria attached on PDA coated surfaces. Above all, these results illustrate that the PDA coating could effectively prevent the bacterial adhesion on membrane surface, and the Ag NPs generation could successfully deactivate the attached bacteria under static conditions.

**Membrane Biofouling Mitigation by Ag NPs during Forward-Osmosis Operation.** Membrane biofouling is a dynamic process involving the continuous adsorption of microorganisms to membrane surfaces and their growth and development into biofilm. The anti-biofouling performances of the modified TFC FO membranes were also investigated in...
the dynamic biofouling experiments. Because both the polyamide surface and polysulfone support were modified with PDA and Ag NPs, it provides a unique opportunity to evaluate their anti-biofouling properties running on both the FO mode and the PRO mode. Figure 3a shows that a gradual yet continuous decline in permeate water flux was observed for all membranes under the PRO mode due to biofouling. Permeate water flux for the pristine membrane declined by 56.4% during 24 h of operation. Under the same conditions, permeate water flux declined for the PDA-coated membranes and the Ag NPs generated membranes by 56.6% and 21.9%, respectively. Conducting the biofouling experiment under the FO mode resulted in an 8.5% decline in the permeate water flux of the pristine membrane (Figure 3b). The PDA-coated membranes exhibited similar water flux decline behavior as the pristine membrane during the biofouling experiments. Importantly, the Ag-NP-generated membranes exhibited only 0.5% permeate water flux decline and presented great biofouling mitigation performances. Even Ag-NP-generated TFC membranes have also PDA coating, and all of the above results illustrate that only Ag NPs rather than PDA coating can substantially suppress the membrane permeate water flux decline and mitigate membrane biofouling.

Moreover, the results shown in Figure 3a,b also indicate that the same TFC FO membrane suffered more-severe water flux decline when running in the PRO mode than in the FO mode. This is because the bottom support layer of TFC membranes has numerous large pores that is more prone to be clogged with foulants inside. In contrast, bacteria can deposit and develop into biofilm only on the surface of top selective layer due to the dense structure of polyamide. Moreover, the TFC FO membrane suffers more-severe fouling-enhanced internal
membranes after 24 h of cross-flow running in the FO system (Ag NPs-T); and (c) the corresponding bacterial culture plate photographs.

Figure 5. Stability and activity of Ag NPs on TFC FO membrane. (a) Residual silver amount on TFC FO membranes after the cross-flow test in the FO system; (b) antimicrobial efficiency of pristine membranes, PDA-coated membranes, Ag-NP-generated membranes, and Ag-NP-generated membranes after 24 h of cross-flow running in the FO system (Ag NPs-T); and (c) the corresponding bacterial culture plate photographs.

Concentration polarization in the PRO mode was lower than 175 L m⁻² (Figure 3b) compared to the pristine membrane, which indicates that the passive moieties have shown some effective anti-biofouling performances. This is because that the experimental conditions favored foulant—membrane interaction, and thus, the passive moieties on membrane surface mitigated the biofouling during the initial biofouling stage. With the progress of biofouling and the development of biofilm, PDA lost its antifouling properties because of its inability to inhibit biofilm development because the experimental conditions favored the foulant—foulant interaction and the PDA coating had no impact on the deposition of bacteria when the membrane surface was fully covered and masked by biofilm. The transition from the stage of the foulant—membrane interaction to that of foulant—foulant dominant interaction was not observed in the PRO mode, which is likely due to the severe fouling conditions in the PRO mode. In addition, a similar outcome was found for other ultrafiltration and nanofiltration membranes modified with hydrophilic polymers such as PDA, poly(ethylene glycol), or heparin; the static anti-adhesion properties of hydrophilic membrane surfaces did not contribute to the enhancement of anti-biofouling performances.

Ag NPs Impairing of Biofilm Growth on TFC FO Membrane Surfaces. To investigate the biofouling mitigation mechanisms of modified TFC membranes, the formed biofilms were analyzed using CLSM. The results shown in Figure 4 indicate that dead cells were more-abundant, while live cells and EPS were less-abundant on the Ag-NP-generated membranes than on the PDA-coated and pristine TFC membranes. Moreover, the top layer of the biofilm formed on the Ag-NP-generated membrane was mainly composed of dead cells, indicating that silver ions leached and diffused to the upper layers of biofilm and broke cell integrity. The image analysis of 3D and the side view of the biofilm further confirms that the biofilm formed on the Ag-NP-generated membrane was thinner and mainly composed of more dead cells with fewer live cells and EPS than biofilms formed on both the PDA-coated and the pristine membrane surfaces. These results indicate that the antimicrobial activity rendered by Ag NPs inhibited the development of biofilm on membrane surfaces.

Ag NPs Impairing of Biofilm Growth on TFC FO Membrane. The stability and activity of Ag NPs on TFC FO membrane was investigated during the dynamic cross-flow operation. We quantified the silver content on membranes with ICP-MS and found that the silver content decreased by 6% after 12 h of operation and further decreased by 17% after 24 h of operation (Figure S5). The morphologies of Ag NPs on polysulfone surface before and after 24 h of operation were observed using an SEM. The average diameter of Ag NPs decreases from 28.1 ± 4.7 to 21.9 ± 5.8 nm, and the number of Ag NPs decreases from 141 ± 19 to 100 ± 26 per square micrometer.

Quantitative analysis was performed to investigate the biomass of biofilms. Less TOC and total protein biomass were found on the Ag NPs generated polysulfone support than that on the PDA coated and pristine membranes (Figure S5). In addition, the PDA coating decreased the TOC content by 29% without affecting the protein biomass on membrane surfaces compared to the pristine surface, which may be due to the suppressed adhesion of organic compounds in feed solution without affecting bacterial growth. Above all, it can be concluded that the antimicrobial activity rendered by Ag NPs inhibits the development of biofilm on membrane surfaces and thus mitigates the membrane biofouling. Moreover, the anti-adhesion properties of hydrophilic membrane surfaces with PDA coating can neither prevent biofilm growth nor mitigate membrane biofouling. However, the anti-adhesion properties of PDA coating has a potential benefit of the efficient removal of biofilm from the modified membrane surfaces during the backwash cleaning procedure, which need a future comprehensive study.

Ag NP Stability and Activity on TFC FO Membrane. To investigate the biofouling mitigation mechanisms of modified TFC membranes, the formed biofilms were analyzed using CLSM. The results shown in Figure 4 indicate that dead cells were more-abundant, while live cells and EPS were less-abundant on the Ag-NP-generated membranes than on the PDA-coated and pristine TFC membranes. Moreover, the top layer of the biofilm formed on the Ag-NP-generated membrane was mainly composed of dead cells, indicating that silver ions leached and diffused to the upper layers of biofilm and broke cell integrity. The image analysis of 3D and the side view of the biofilm further confirms that the biofilm formed on the Ag-NP-generated membrane was thinner and mainly composed of more dead cells with fewer live cells and EPS than biofilms formed on both the PDA-coated and the pristine membrane surfaces. These results indicate that the antimicrobial activity rendered by Ag NPs inhibited the development of biofilm on membrane surfaces.

Quantitative analysis was performed to investigate the biomass of biofilms. Less TOC and total protein biomass were found on the Ag NPs generated polysulfone support than that on the PDA coated and pristine membranes (Figure S5). In addition, the PDA coating decreased the TOC content by 29% without affecting the protein biomass on membrane surfaces compared to the pristine surface, which may be due to the suppressed adhesion of organic compounds in feed solution without affecting bacterial growth. Above all, it can be concluded that the antimicrobial activity rendered by Ag NPs inhibits the development of biofilm on membrane surfaces and thus mitigates the membrane biofouling. Moreover, the anti-adhesion properties of hydrophilic membrane surfaces with PDA coating can neither prevent biofilm growth nor mitigate membrane biofouling. However, the anti-adhesion properties of PDA coating has a potential benefit of the efficient removal of biofilm from the modified membrane surfaces during the backwash cleaning procedure, which need a future comprehensive study.

Ag NP Stability and Activity on TFC FO Membrane. To investigate the biofouling mitigation mechanisms of modified TFC membranes, the formed biofilms were analyzed using CLSM. The results shown in Figure 4 indicate that dead cells were more-abundant, while live cells and EPS were less-abundant on the Ag-NP-generated membranes than on the PDA-coated and pristine TFC membranes. Moreover, the top layer of the biofilm formed on the Ag-NP-generated membrane was mainly composed of dead cells, indicating that silver ions leached and diffused to the upper layers of biofilm and broke cell integrity. The image analysis of 3D and the side view of the biofilm further confirms that the biofilm formed on the Ag-NP-generated membrane was thinner and mainly composed of more dead cells with fewer live cells and EPS than biofilms formed on both the PDA-coated and the pristine membrane surfaces. These results indicate that the antimicrobial activity rendered by Ag NPs inhibited the development of biofilm on membrane surfaces.

Quantitative analysis was performed to investigate the biomass of biofilms. Less TOC and total protein biomass were found on the Ag NPs generated polysulfone support than that on the PDA coated and pristine membranes (Figure S5). In addition, the PDA coating decreased the TOC content by 29% without affecting the protein biomass on membrane surfaces compared to the pristine surface, which may be due to the suppressed adhesion of organic compounds in feed solution without affecting bacterial growth. Above all, it can be concluded that the antimicrobial activity rendered by Ag NPs inhibits the development of biofilm on membrane surfaces and thus mitigates the membrane biofouling. Moreover, the anti-adhesion properties of hydrophilic membrane surfaces with PDA coating can neither prevent biofilm growth nor mitigate membrane biofouling. However, the anti-adhesion properties of PDA coating has a potential benefit of the efficient removal of biofilm from the modified membrane surfaces during the backwash cleaning procedure, which need a future comprehensive study.
Therefore, the silver leaching comes from both losing particles and the dissolution of the Ag NPs. The antimicrobial efficiency of TFC FO membrane was investigated using the CFU counting method. Results shown in Figure S6.c present that the numbers of attached live P. aeruginosa decreased by 33.3 ± 4.2% and 97.0 ± 2.4% for the PDA-coated and the Ag-NP-generated membranes compared to the pristine membrane, agreeing with the results of CLSM image analysis in the static condition. The Ag NPs generated membranes after 24 h of cross-flow running in the FO system (Ag NPs-T) still exhibited great antimicrobial efficiency of 96.1%, similar to that of the fresh Ag-NP-generated membrane. Therefore, the Ag NPs are stable and active on TFC FO membrane and hold great potential for sustainable antibacterial performance.

**Implications for Biofouling Mitigation of TFC FO Membrane.** In this study, we fabricated TFC FO membranes modified with both PDA as the passive moiety and Ag NPs as the active moiety and investigated the performances and mechanisms of their biofouling mitigation in the FO process. The PDA coating on TFC FO membranes exhibits passive anti-adhesive properties owing to hydrophilicity in static conditions but shows little effect on biofouling mitigation in the dynamic cross-flow conditions due to its inability to inactivate bacteria growth. The Ag NPs present active antibacterial properties in both the static and the dynamic conditions. The biofilm analysis illustrates that the mechanism of biofouling mitigation by Ag NPs is the prevention of biofilm growth and the reduction of microbial biomass through the inactivation of the attached bacteria. In addition, TFC FO membranes suffered more-severe water flux decline from biofouling when running in the PRO mode than in the FO mode. More importantly, Ag NPs show good stability and activity on the TFC FO membrane surfaces during 24 h of cross-flow operation. These findings provide new insights on the effective surface engineering of FO membranes with active moieties for efficient biofouling mitigation. Future investigations should focus on the long-term anti-biofouling properties of TFC FO membranes with the passive and active moieties when coupled with cleaning strategies to treat complex water systems.

**ACKNOWLEDGMENTS**

The authors gratefully acknowledge the funding support from National Natural Science Foundation of China (nos. 21476249 and 51708408), the Chang-Jiang Scholars and Innovative Research Team in the University of Ministry of Education, China (no. IRT-17R80), Program for Innovative Research Team in University of Tianjin (no. TD13-5044), and the Science and Technology Plans of Tianjin (no. 17PTSYJC00060).

**REFERENCES**

1. Vorosmarty, C. J.; Green, P.; Salisbury, J.; Lammers, R. B. Global water resources: vulnerability from climate change and population growth. *Science 2000*, 289 (5477), 284–288.
2. Shannon, M. A.; Bohn, P. W.; Elimelech, M.; Georgiadis, J. G.; Marinas, B. J.; Mayes, A. M. Science and technology for water purification in the coming decades. *Nature* 2008, 452 (7185), 301–310.
3. Elimelech, M.; Phillip, W. A. The future of seawater desalination: energy, technology, and the environment. *Science* 2011, 333 (6043), 712–717.
4. Shaffer, D. L.; Werber, J. R.; Jaramillo, H.; Lin, S.; Elimelech, M. Forward osmosis: Where are we now? *Desalination* 2015, 356, 271–284.
5. Lutchmiah, K.; Verlieide, A. R.; Roest, K.; Rietveld, L. C.; Cornelissen, E. R. Forward osmosis for application in wastewater treatment: a review. *Water Res.* 2014, 58 (3), 179–197.
6. Cornelissen, E. R.; Harnsen, D.; de Korte, K. F.; Ruiken, C. J.; Qin, J. J.; Oo, H.; Wessels, L. P. Membrane fouling and process performance of forward osmosis membranes on activated sludge. *J. Membr. Sci.* 2008, 319 (1–2), 158–168.
7. Wintgens, T.; Melin, T.; Schäfer, A.; Khan, S.; Muston, M.; Bixio, D.; Thoeyle, C. The role of membrane processes in municipal wastewater reclamation and reuse. *Desalination 2005*, 178 (1), 1–11.
8. Han, G.; Zhang, S.; Li, X.; Chung, T. S. High performance thin film composite pressure retarded osmosis (PRO) membranes for renewable salinity-gradient energy generation. *J. Membr. Sci.* 2013, 440 (8), 108–121.
9. Logan, B. E.; Elimelech, M. Membrane-based processes for sustainable power generation using water. *Nature* 2012, 488 (7411), 313–319.
10. Yip, N. Y.; Tiraferrì, A.; Phillip, W. A.; Schiffman, J. D.; Elimelech, M. High performance thin-film composite forward osmosis membrane. *Environ. Sci. Technol.* 2010, 44 (10), 3812–3818.
11. Han, G.; Zhao, B.; Fu, F.; Chung, T. S.; Weber, M.; Staudt, C.; Maletzko, C. High performance thin-film composite membranes with mesh-reinforced hydrophilic sulfonated polyphenylenesulfone (sPPSU) substrates for osmotically driven processes. *J. Membr. Sci.* 2016, 502, 84–93.
12. Ren, J.; Mccutcheon, J. R. A new commercial thin film composite membrane for forward osmosis. *Desalination* 2014, 343 (12), 187–193.
13. Barzeev, E.; Perreault, F.; Straub, A. P.; Elimelech, M. Impaired performance of pressure-retarded osmosis due to irreversible biofouling. *Environ. Sci. Technol.* 2015, 49 (21), 13050–13058.
14. Youn, H.; Baek, Y.; Yu, J.; Youn, J. Biofouling occurrence process and its control in the forward osmosis. *Desalination 2013*, 325 (20), 30–36.
15. Lee, S.; Boo, C.; Elimelech, M.; Hong, S. Comparison of fouling behavior in forward osmosis (FO) and reverse osmosis (RO). *J. Membr. Sci.* 2010, 365 (1), 34–39.
16. Zhang, R.; Liu, Y.; He, M.; Su, Y.; Zhao, X.; Elimelech, M.; Jiang, Z. Antifouling membranes for sustainable water purification: strategies and mechanisms. *Chem. Soc. Rev.* 2016, 45 (21), 5888–5924.
membranes for osmotic power generation. Antifouling polymer brushes. Polyamide membranes modified with biocidal nanoparticles and Nielsen, M.; Elimelech, M. Control of biofouling on reverse osmosis. *Environmental Science & Technology* 2014, 48 (16), 9898–9907.

- Proteins, bacteria, and marine organisms.

- Modified polymer membranes for biofouling mitigation. *Environ. Sci. Technol.* 2015, 49, 15566.

- Tang, C. Y.; Kwon, Y. N.; Leckie, J. O. The role of foulant-foam electrostatic interaction on limiting flux for RO and NF membranes during humic acid fouling-Theoretical basis, experimental evidence, and AFM interaction force measurement. *J. Membr. Sci.* 2009, 326 (2), 526–532.

- Liu, X.; Foo, L. X.; Li, Y.; Lee, J. Y.; Cao, B.; Tang, C. Y. Fabrication and characterization of nanocomposite pressure retarded osmosis (PRO) membranes with excellent anti-biofouling property and enhanced water permeability. *Desalination* 2016, 356, 187–207.

- Zhu, L. J.; Zhu, L. P.; Zhao, Y. F.; Zhu, B. K.; Xu, Y. Y. Anti-fouling and anti-bacterial polyethersulfone membranes quantized from the additive of poly(2-dimethylamino ethyl methacrylate) grafted SiO2 nanoparticles. *J. Membr. Sci.* 2014, 421, 15566.

- Tang, C. Y.; Kwon, Y. N.; Leckie, J. O. The role of foulant-foam electrostatic interaction on limiting flux for RO and NF membranes during humic acid fouling-Theoretical basis, experimental evidence, and AFM interaction force measurement. *J. Membr. Sci.* 2009, 326 (2), 526–532.

- Liu, X.; Foo, L. X.; Li, Y.; Lee, J. Y.; Cao, B.; Tang, C. Y. Fabrication and characterization of nanocomposite pressure retarded osmosis (PRO) membranes with excellent anti-biofouling property and enhanced water permeability. *Desalination* 2016, 356, 137–148.

- Wang, J.; Wang, Y.; Zhang, Y.; Uliana, A. A.; Zhu, J.; Liu, J.; van der Bruggen, B. Zeolitic imidazolate framework/graphene oxide hybrid nanosheets functionalized thin film nanocomposite membrane for enhanced antimicrobial performance. *ACS Appl. Mater. Interfaces* 2016, 8 (38), 25508–25519.

- Ben-Sasson, M.; Zodrow, K. R.; Genggeng, Q.; Kang, Y.; Giannelis, E. P.; Elimelech, M. Surface functionalization of thin-film composite membranes with copper nanoparticles for antimicrobial surface properties. *Environ. Sci. Technol.* 2014, 48 (1), 384–393.

- She, Q.; Wang, R.; Fan, A. G.; Tang, C. Y. Membrane fouling in osmotically driven membrane processes: A review. *J. Membr. Sci.* 2016, 499, 201–233.

- Kang, S. T.; Subramani, A.; Hoek, E. M. V.; Deshusses, M. A.; Matsumoto, M. R. Direct observation of biofouling in cross-flow microfiltration: mechanisms of deposition and release. *J. Membr. Sci.* 2004, 244 (1–2), 151–165.

- Liu, C. X.; Zhang, D. R.; He, Y.; Zhao, X. S.; Bai, R. Modification of membrane surface for anti-biofouling performance: Effect of anti-adhesion and anti-bacteria approaches. *J. Membr. Sci.* 2010, 346 (1), 121–130.

- Lee, H.; Dellatore, S. M.; Miller, W. M.; Messersmith, P. B. Mussel-inspired surface chemistry for multifunctional coatings. *Science* 2007, 318 (5849), 426–430.

- Lee, H.; Rho, J.; Messersmith, P. B. Facile conjugation of biomolecules onto surfaces via mussel adhesive protein inspired coatings. *Adv. Mater.* 2009, 21 (4), 431–43.

- Liu, Z.; Hu, Y. Sustainable antibiofouling properties of thin film composite forward osmosis membrane with rechargeable silver nanoparticles loading. *ACS Appl. Mater. Interfaces* 2016, 8 (33), 21666–21673.

- Bernsman, F.; Ball, V.; Addiego, F.; Ponche, A.; Michel, M.; Gracio, J. J.; Di D. A.; Toniazzo, V.; Ruch, D. Dopamine-melanin film deposition depends on the used oxidant and buffer solution. *Langmuir* 2011, 27 (6), 2819–2825.

- Karkhaneci, H.; Takagi, R.; Matsuyama, H. Biofouling resistance of reverse osmosis membrane modified with polydopamine. *Desalination* 2014, 336 (1), 87–96.

- Mcclower, B. D.; Park, H. B.; Ju, H.; Rowe, B. W.; Miller, D. J.; Chun, B. J.; Kin, K.; Freeman, B. D. Influence of polydopamine deposition conditions on pure water flux and fouling adhesion resistance of reverse osmosis, ultrafiltration, and microfiltration membranes. *Polymer* 2010, 51 (15), 3472–3485.

- Tang, L.; Livi, K. J. T.; Chen, K. L. Polysulfone membranes modified with bioinspired polydopamine and silver nanoparticles formed in situ to mitigate biofouling. *Environ. Sci. Technol. Lett.* 2015, 2 (3), 59–65.

- Ben-Sasson, M.; Lu, X.; Bar-Zeev, E.; Zodrow, K. R.; Nejati, S.; Qi, G.; Giannelis, E. P.; Elimelech, M. In situ formation of silver nanoparticles on thin-film composite reverse osmosis membranes for biofouling mitigation. *Water Res.* 2014, 62 (10), 260–270.

- Yang, Z.; Wu, Y.; Wang, J.; Cao, B.; Tang, C. Y. In situ reduction of silver by polydopamine: A novel antimicrobial modification of a thin-film composite polyeleimide membrane. *Environ. Sci. Technol.* 2016, 50 (17), 9543–9550.

- Liu, Z.; An, X.; Dong, C.; Zheng, S.; Mi, B.; Hu, Y. Modification of thin film composite polyeleimide membranes with 3D hyperbranched polyglycerol for simultaneous improvement in their filtration performance and antifouling properties. *J. Membr. Chem.* 2017, 5 (4), 23190–23197.

- Xie, M.; Bar-Zeev, E.; Hashmi, S. M.; Nghiem, L. D.; Elimelech, M. Role of reverse divalent cation diffusion in forward osmosis biofouling. *Environ. Sci. Technol.* 2015, 49 (22), 13222–13229.

- Smith, P. K.; Krohn, R. I.; Hermanson, G. T.; Mallia, A. K.; Gartner, F. H.; Provenzano, M. D.; Fujimoto, E. K.; Goekce, N. M.; Olson, B. J.; Klenk, D. C. Measurement of protein using bicinchoninic acid. *Anal. Biochem.* 1985, 150 (1), 76–85.

- Zhou, J.; Shi, Q.; Li, X.; Tang, C. Y.; Hu, Y.; Shai, H.; Jin, J.; Li, X.; Yin, J.; Stagnaro, P. Improved biocompatibility and antifouling property of polypropylene non- woven fabric membrane by surface grafting zwitterionic polymer. *J. Membr. Sci.* 2011, 369 (1–2), S1–S12.

- Jiang, J.; Zhu, L.; Zhu, L.; Zhou, Z.; Xu, Y. Surface characteristics of a self-polymerized dopamine coating deposited on hydrophobic polymer films. *Langmuir* 2011, 27 (23), 14180–14187.

- Huang, L.; Zhao, S.; Wang, Z.; Wu, J.; Wang, J.; Wang, S. In situ immobilization of silver nanoparticles for improving permeability, antifouling and anti-bacterial properties of ultrafiltration membrane. *J. Membr. Sci.* 2016, 499, 269–281.

- Yin, J.; Deng, B. Polymer-matrix nanocomposite membranes for water treatment. *J. Membr. Sci.* 2015, 479, 256–275.

- Tiraferri, A.; Yip, N. Y.; Straub, A. P.; Castrillon, S. R.-V.; Elimelech, M. A method for the simultaneous determination of transport and structural parameters of forward osmosis membranes. *J. Membr. Sci.* 2013, 444 (1), 523–538.

- Su, L.; Yu, Y.; Zhao, Y.; Liang, F.; Zhang, X. Strong antibacterial polydopamine coatings prepared by a shaking-assisted method. *Sci. Rep.* 2016, 6, 24420.

- Yu, B.; Liu, J.; Liu, S.; Zhou, F. Pdop layer exhibiting zwitterionicity: a simple electrochemical interface for governing ion permeability. *Chem. Commun.* 2016, 52 (32), 5900–5902.

- Van Houdt, R.; Michiels, C. W. Role of bacterial cell surface structures in Escherichia coli biofilm formation. *Res. Microbiol.* 2005, 156 (5–6), 626–633.

- Miller, D. J.; Araújo, P. A.; Correia, P. B.; Ramsey, M. M.; Kruthof, J. C.; van Loosdrecht, M. C. M.; Freeman, B. D.; Paul, D. R.
Whiteley, M.; Vrouwenvelder, J. S. Short-term adhesion and long-term biofouling testing of polydopamine and poly(ethylene glycol) surface modifications of membranes and feed spacers for biofouling control. *Water Res.* 2012, 46 (12), 3737–3753.

(54) Marambio-Jones, C.; Hoek, E. M. V. A review of the antibacterial effects of silver nanomaterials and potential implications for human health and the environment. *J. Nanopart. Res.* 2010, 12 (5), 1531–1551.