Engineering of functionalized carbon nano-onions reinforced nanocomposites: Fabrication, biocompatibility, and mechanical properties

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Poly 4-mercaptophenyl methacrylate-carbon nano-onions ((PMPMA-CNOs = f-CNOs) were reinforced with polycaprolactone (PCL) to produce PCL/f-CNO nanocomposites using probe sonication. The physicochemical properties of nanocomposites were systematically studied to analyze cell viability and proliferation. In vitro cytotoxicity of PCL/f-CNO nanocomposites was measured with osteoblast cells, and improved cell viability was observed. The cytotoxicity of f-CNOs to osteoblasts was dose-dependent, and PCL/f-CNO (0.5 wt%) nanocomposites showed more than 90% of viability as compared to pristine PCL. Similarly, PCL/f-CNO (0.5 wt%) nanocomposites showed substantial enhancement in mechanical properties. The yield strength, tensile strength, Young modulus, elastic modulus, and fracture toughness were also upgraded at high content of f-CNOs (0.5 wt%). The concentration of f-CNOs considerably influenced the strengthening of PCL/f-CNO nanocomposites, which shows its degree of colloidal dispersion stability and extent of polymer wrapping within the PCL matrix. Nevertheless, these nontoxic PCL/f-CNO nanocomposites can be used as promising biomaterials for orthopedic applications.

Introduction

Biodegradable medical-grade biomaterials have received immense attention in recent times due to their potential applications as cardiovascular interventional devices and orthopedic implants [1, 2]. The most commonly used materials for bone fracture fixation are 316L stainless steel, cobalt–chromium-based alloys, titanium alloys, and various polymers [3, 4]. The foremost desirable characteristic of implantable materials is their degradability after the bone has healed. However, the above metals are nondegradable, and cause the risk of local inflammation and, hence, premature replacement of implants [5]. Therefore, biodegradable polymers have been investigated to reduce such complications [6]. In this context, polymer-based biodegradable biomaterials including poly(lactic-co-glycolic acid)
(PLGA), poly(L-lactic acid) (PLA), and polycaprolactone (PCL) have been extensively used in clinics to benefit with the renovation of human tissues, for instance as absorbable bone plates, pins, and bone screws [7, 8]. Among these polymers, PCL is a promising biodegradable polymer due to its great biocompatibility, significant toughness, and outstanding tensile properties (80% of elongation at a breakpoint) [5, 9]. PCL is a semicrystalline polymer that has higher fracture energy and a sluggish degradation rate than most biopolymers [9, 10]. In addition, the hydrophobic nature of PCL is more appropriate for a coating applied on metal alloys [5, 11]. However, PCL has some detriments when used alone due to a lack of desired biocompatibility and insufficient strength [7, 11]. Therefore, it is important to overcome these drawbacks, which can be accomplished by using appropriate second-phase reinforcement. For instance, most of the biopolymers including PCL are reinforced with nanofillers, such as graphite (Gr), carbon fibers (CF), graphite fluoride (CFx), carbon nanotubes (CNTs), graphene oxide (GO), and hydroxyapatite (HA), to improve their mechanical properties and biocompatibility [9, 11, 12, 13, 14].

Among carbon fillers, CNTs are considered to be ideal reinforcing nanofillers for polymer matrices because of their unique structure and properties [15]. However, CNTs display different levels of toxicity based on their shape, size, surface area to volume ratio, synthetic methods, and extent of oxidation [13, 16]. Besides, CNTs can extend their toxicity through protein synthesis, oxidative stress, mitochondrial amendments, intramolecular metabolic paths, apoptosis, and necrosis [17]. However, graphene and its derivative affect the plasma membrane, membrane organelles, and cytoskeleton (e.g., lysosomes, nucleus, and mitochondrion) [18]. Therefore, it is highly appropriate to develop a facile method to fabricate nontoxic carbon-based nanocomposites with improved mechanical properties and biocompatibility.

Carbon nano-onions (CNOs) are a new member of carbonaceous materials, which are first described by Ugarte in 1992 [19]. Because of their inimitable physicochemical properties, CNOs have been used in numerous research fields, including catalysis, lithium-ion battery, supercapacitor, therapeutic, cell imaging, diagnostic, and biomedical applications [20, 21, 22, 23, 24, 25]. The toxicity of CNOs was systematically measured and they were found to be more biocompatible than MWCNTs with benign inflammatory response [24]. Also, pristine CNOs, oxidized CNOs, and PEGylated CNOs were found to be nontoxic and showed excellent cell viability with fibroblasts [26].

Recently, poly 4-mercaptophenyl methacrylate-CNOs (PMPMA-CNOs = f-CNOs) were synthesized and reinforced with ultra-high molecular weight polyethylene (UHMWPE) to engineer UHMWPE/f-CNO nanocomposites [27]. The physicochemical properties and cytotoxicity of UHMWPE/f-CNO nanocomposites were systematically studied. These composites exhibited good cell growth, adhesion, and viability with osteoblast cells. Besides, the nanocomposites showed improved tensile strength (83%) compared to pure UHMWPE. In addition, f-CNOs were reinforced with zein protein to develop zein/f-CNO composite hydrogels for pH-responsive controlled drug release [28]. The zein/f-CNO composite hydrogels showed improved mechanical properties and cytocompatibility [28]. These results of CNOs motivated us to fabricate PCL/CNO nanocomposites as alternate biomaterials for orthopedic applications.

Nevertheless, to the best of our knowledge, the development of f-CNO reinforced PCL/f-CNO nanocomposites has not yet been reported. Consequently, it was hypothesized that f-CNOs could be uniformly dispersed and reinforced homogeneously within the PCL matrix. Good colloidal dispersion of f-CNO particles can improve the tensile properties, biodegradability, and biocompatibility of PCL/f-CNO nanocomposites.

In the current study, f-CNO reinforced PCL/f-CNO nanocomposites were manufactured using probe sonication followed by hydraulic pressing. Two different weight percentages (0.2 and 0.5 wt%) of f-CNOs were used to develop PCL/f-CNO nanocomposites. Considering the high surface area of CNOs, the weight percentage of f-CNOs was kept very low. Thermal, mechanical, and cytotoxicity assessments of PCL/f-CNO nanocomposites were explored systematically. Young's modulus, tensile strength, elongation at the break, and crystallinity of nanocomposites were improved compared to pristine PCL. Osteoblast cells were used to measure the cytotoxicity of PCL/f-CNO nanocomposites. The in vitro biocompatibility measurements reveal that cell viability, adhesion, and cell growth were improved at 0.5 wt% of f-CNOs, which signifies a positive effect of f-CNOs on osteoblast cells.

**Results and discussion**

To achieve good physicochemical properties of f-CNO–based nanocomposites, it is primarily vital to disperse the CNO particles throughout the polymer matrix uniformly, which is achieved in the current study by using the probe sonication (sonochemistry) method. Probe sonication is a simple, easy, and short-time physicochemical method that depends on high-intensity ultrasound [32, 33]. Ultra probe-sonication depends on the acoustic cavitation phenomenon. Acoustic cavitation occurs in the liquid environment due to the fluctuation of pressure, which generates bubble growth, followed by bubble collapse and internal turbulence. Finally, this ultrasound energy converts into high temperature and confined pressure. Ultrasound waves exchange through carbon nanoparticles (CNOs), which are held by Van der Waals forces, and/or π–π stacking.
In this method, CNOs are arranged in a liquid environment. Colloidal stabilization of CNOs is achieved by the physisorption of PMPMA chains onto the CNO surface. Therefore, probe sonication would be a good method to produce uniformly dispersed and stabilized CNOs in aqueous environments or organic solvents. These colloidal stabilized CNOs are highly efficient to upgrade the electrical, tensile, and elasticity properties of the nanocomposites. Figure 1 shows the fabrication method of the PCL/f-CNO nanocomposite sheet and osteoblast cell growth. Initially, pure f-CNOs and PCL were probe-sonicated to obtain a uniform suspension of PCL/f-CNO composites. The resulted black mixture was oven-dried for 12 h. The dried black mixture was subjected to hydraulic pressing to provide PCL/f-CNO nanocomposite sheets (Fig. 1). In addition, live/dead cell growth was studied with osteoblast cells, and more than 98% of live cells were observed (Fig. 1). This indicates the good biocompatibility of functionalized CNOs to the bone cells. Biocompatibility is the primary characteristic of medical grade materials for orthopedic implantation.

The microstructure of fractured PCL/f-CNO nanocomposites was measured using scanning electron microscopy (SEM) analysis, and the SEM images are illustrated in Fig. 2. As shown in Fig. 2(a), the fracture surface of the PCL/f-CNOs (0.2 wt%) displays irregular surfaces. It shows that f-CNOs are uniformly distributed within the PCL matrix. Also, PCL/f-CNO (0.5 wt%) composites display a similar asymmetrical surface. Thus, the SEM analysis reveals that f-CNOs are homogeneously dispersed and reinforced within the PCL chains. The uniform dispersion can enhance the mechanical and crystallinity properties of PCL. It is worth mentioning that the homogeneous dispersion of CNOs and HA nanoparticles throughout the polymer matrix could enrich the crystallinity and mechanical properties of polymers [27, 28, 34].

Figure 3(a) illustrates the roughness (Ra) measurements of pure PCL and PCL/f-CNO nanocomposites. As shown in Fig. 3(a), PCL/f-CNO nanocomposites exhibited improved roughness compared to pristine PCL. Particularly, PCL/f-CNOs (0.2 wt%) showed 0.21 μm, and PCL/f-CNOs (0.5 wt%) exhibited 0.38 μm of roughness, whereas pure PCL revealed 0.12 μm of roughness. Overall, PCL/f-CNOs (0.5 wt%) exhibited the highest roughness values, which signifies the original flat surface of pristine PCL was interrupted by the inclusion of f-CNOs. The improved roughness of nanocomposites can trigger cell adhesion and cell growth on the surface of nanocomposites [13, 27, 28].

Table I presents the yield strength, tensile strength, elongation at the break, and Young modulus of pure PCL and PCL/f-CNO nanocomposites. Figure 3(b) and Table I unveil the enhancement in the mechanical properties of PCL with the inclusion of f-CNOs. The Young modulus of pure PCL, PCL/f-CNOs (0.2 wt%), and PCL/f-CNOs (0.5 wt%) are calculated to be 738.90, 975.77, and 1294.54 MPa, respectively. Similarly, yield strength also improved with the addition of f-CNOs, particularly 12, 18, and 25 MPa of yield strength was observed in pure PCL, PCL/f-CNOs (0.2 wt%), and PCL/f-CNOs (0.5 wt%), respectively. The maximum enhancement in tensile strength up to 84 MPa was attained with 0.5 wt% of f-CNO content, as compared to 45 MPa in PCL. The elongation at break of composite samples was slightly provoked by the addition of f-CNOs, as shown in Table I. The tensile results demonstrated that 2.1-fold of stiffness was upgraded in PCL/f-CNO (0.5 wt%) nanocomposites compared to pure PCL. Overall, the inclusion of f-CNOs enriched the mechanical properties of PCL. It was presumed that due to the tensile force, pure PCL and f-CNOs strained to provide the shuffled alignment. The improved tensile properties demonstrate the uniform dispersion of f-CNOs within the PCL matrix. Besides, during the tensile test, external force might have been transferred between f-CNOs and the PCL matrix. This phenomenon could be due to the Van der Waals attractions or π–π interactions between f-CNOs and PCL chains. Overall, the mechanical properties of nanocomposites revealed that with the inclusion of f-CNOs, the tensile properties of PCL/f-CNO composites have improved, which is one of the basic and vital characteristics of orthopedic devices. The mechanical results were consistent with the previous report [27].
The thermal stability and decomposition behavior of neat PCL and PCL/f-CNO nanocomposites were measured using thermal gravimetric analysis (TGA). The initial temperature ($T_{0.1}$) of onset degradation and the midpoint degradation temperature ($T_{0.5}$) of weight loss of specimens are presented in Fig. 4(a). The PCL/f-CNO nanocomposites and pure PCL samples showed the thermal decomposition in the temperature range 345–462 °C. As shown in Fig. 4(a), pristine PCL showed $T_{0.1}$ and $T_{0.5}$ approximately at 354 °C and 368 °C, respectively. With the inclusion of f-CNOs, the PCL/f-CNO nanocomposites showed increased onset degradation temperature. Specifically, PCL/f-CNO nanocomposites showed weight loss in the temperature range 370–460 °C. Thermal degradation of the pure polymer occurred readily when the molecular chains had high mobility at the interface. However, thermal degradation of pure PCL was reduced with the inclusion of f-CNOs at the interface. This could be due to the mobility restriction of PCL chains at the interface. Overall, the TGA results reveal that the thermal stability of the nanocomposites was improved with the inclusion of f-CNOs.

The DSC measurements can predict the polymer characteristics, including the melting temperature, degree of crystallinity, miscibility, and intermolecular interactions [13, 27]. Therefore, pure PCL and PCL/f-CNO nanocomposites were subjected to the DSC analysis, and the first heating, as well as cooling curves of all the samples, are illustrated in Fig. 4(b). The DSC results showed that the crystallinity of the nanocomposites was enhanced in the presence of f-CNOs. The melting temperature of pristine PCL and PCL/f-CNOs was in the range of 57–61 °C. Generally, carbonaceous nanocomposites are a mixture of polymer chains and carbon materials that are not covalently attached. The nanocomposites display compatibility due to hydrogen bonding, dipole–dipole interactions, and other forces between the functional groups of the nanocomposites. Specifically, neat PCL exhibited the crystallization temperature ($T_c$) and melting temperature ($T_m$) approximately at 25.0 °C and 57.0 °C, respectively, whereas PCL/f-CNO (0.2 wt%) nanocomposites showed ($T_c$) and ($T_m$) at around 26.7 °C and 58.1 °C, respectively. Besides, PCL/f-CNO (0.5 wt%) nanocomposites presented ($T_c$) and ($T_m$) at approximately 26.4 °C and 60.1 °C, respectively. These DSC values are in agreement with the previous results [29].

The cell viability, cellular morphology, and LIVE/DEAD cell imaging of PCL/f-CNO nanocomposites were measured to check the cytotoxic influence of f-CNOs on the biocompatibility of nanocomposites. For this, human osteoblast cells were utilized to measure the cell viability of nanocomposites, and the results are presented in Fig. 5(a). The cells were cultured on the surface of neat PCL and PCL/f-CNO nanocomposites. As shown in Fig. 5(a), the cell viability of the nanocomposites was measured and compared with a control (tissue culture plate; TC). All the composite specimens exhibited approximately 70% of cell adhesion at day one post-seeding, whereas the TC plate showed around 95% of cell adhesion. On day one, neat PCL and PCL/f-CNO nanocomposites showed a similar cell adhesion. The cell viability was prominently influenced by cell adhesion, and the TC plate exhibited a somewhat higher cell number on day three. However, after seven and fourteen days of incubation, the cell viability was significantly enhanced.
improved in PCL/f-CNO nanocomposites. Besides, PCL/f-CNO nanocomposites displayed a slightly higher proliferation rate than the control. Predominantly, PCL/f-CNO (0.5 wt%) nanocomposites exhibited a 27% higher proliferation rate than the control for over fourteen days of study, which specifies a positive effect of f-CNOs on osteoblast cells. Thus, the cell viability results revealed that the PCL/f-CNO nanocomposites could support improved cell growth and have a positive influence on biocompatibility. Furthermore, the viability results also confirmed that conjugated CNOs are more biocompatible toward osteoblasts than CNT-UHMWPE nanocomposites [13].

In addition, osteoblast cell images were also measured using the LIVE/DEAD cell imaging kit. The osteoblast cells were cultured on the surface of nanocomposites for fourteen days, and the images were recorded using a fluorescence microscope [Figs. 5(b)–5(e)]. Red color indicates dead cells and green color indicates live cells. PCL/f-CNO (0.5 wt%) composites showed more than 98% and PCL/f-CNO (0.2 wt%) composites exhibited more than 91% of live cells. Insignificant fractions of dead cells were detected on the nanocomposites. Interestingly, flat- and stretched-shaped cells were observed on the surface of the nanocomposites [Figs. 5(b)–5(e)]. This could be due to the alignment of f-CNOs within the PCL matrix that could regulate the cell attachment. Similar results were observed with human mesenchymal stem cells on the pattern of isolated CNTs [30, 31].

An ideal bone-implant substitute should be appropriate for cell adhesion. The SEM analysis was performed to observe the cellular morphology and cellular adhesions to the PCL/f-CNO (0.5 wt%) nanocomposite scaffold. A typical overview of osteoblast cell morphology on the PCL/f-CNO (0.5 wt%) composite specimen for 14 days is illustrated in Fig. 6a. The osteoblast cells cultured on the PCL/f-CNO sample showed flattened (green arrows) and spindle-shaped (yellow arrows) cells. The cells on the PCL/f-CNO (0.5 wt%) nanocomposites exhibited fewer cytoplasmic extensions, and the cells were elongated in all directions on the composite surface. Pseudopodium-like structures (white arrow) were observed along

**Figure 4:** (a) TGA and (b) DSC curves of neat PCL and PCL/f-CNO nanocomposites, respectively.

**Figure 5:** (a) Cell viability of the tissue culture plate (control), neat PCL, and PCL/f-CNO nanocomposites. Qualitative cell growth of osteoblast cells after fourteen days on (b) tissue culture well (control), (c) pure PCL, (d) PCL/f-CNOs (0.2 wt%), and (e) PCL/f-CNOs (0.5 wt%). The green color shows live cells, and the red color specifies dead cells. The scale bar is 100 μm.
with many thin filopodia (red arrows) attached to the PCL/f-CNO (0.5 wt%) surface. For clarity, we have added the SEM image of the PCL/f-CNO (0.5 wt%) composites without cell culture treatment as a blank sample [Fig. 6(b)] to see the morphological differences between the cell-cultured sample [Fig. 6(a)] and the blank sample. As seen in Fig. 6(b), without cellular growth, the surface of the blank sample was very smooth, which confirmed that the cells were attached to the surface of the PCL/f-CNO composites [Fig. 6(a)]. These results indicated that the PCL/f-CNO (0.5 wt%) composites would be interesting for potential biomedical applications. The cell attachment results were consistent with the previous reports [7, 35, 36, 37].

**Conclusions**

Poly 4-mercaptophenyl methacrylate-carbon nano-onions (f-CNOs) reinforced PCL/f-CNO nanocomposites were successfully fabricated using probe sonication followed by hydraulic pressing. The content of f-CNOs was found to be critical for upgrading mechanical properties. Young’s modulus of PCL was increased from 738.90 to 1294.54 MPa at PCL/f-CNOs (0.5 wt%). A similar trend was noticed for tensile strength also, with a maximum of 80% improvement at 0.5 wt% f-CNO loading. Also, yield strength and elongation at the break of PCL/f-CNO nanocomposites were improved with the inclusion of f-CNOs. The f-CNO concentration along with its colloidal dispersion and matrix reinforcement interfacial bonding was found to activate strengthening mechanisms. The biocompatibility of f-CNO and PCL/f-CNO nanocomposites to osteoblast cells was found to be dose dependent. The PCL/f-CNO (0.5 wt%) nanocomposites exhibited more than 90% of cell viability and higher than 98% of survivability of bone cells on the surface of the PCL/f-CNO (0.5 wt%) nanocomposite. The current study suggests PCL/f-CNO nanocomposites would be considered as promising biomaterials for orthopedic applications and beyond.

**Experimental section**

**Materials and methods**

All the organic solvents and reagents were purchased from commercial suppliers and used without further purification. Polycaprolactone ($M_w \sim 80,000$), $N$-hydroxysuccinimide (NHS), 1-ethyl-3-(3-dimethylaminopropyl) carbodiimide (EDC), 4-(dimethylamino)pyridine (DMAP), and 2,2′-Azobis(2-methylpropionitrile) (AIBN) were purchased from Sigma Aldrich (St. Louis, MO, USA). Osteoblast cells were procured from the American Type Culture Collection (ATCC, Manassas, VA). Dulbecco’s modified Eagle’s medium/F12 without phenol red (DMEM/F12), phosphate-buffered saline (PBS) (pH 7.4), fetal bovine serum (FBS), penicillin/streptomycin, and trypsin were bought from Gibco Invitrogen (CA, USA). CellTiter96® AQueous One Solution Cell Proliferation Assay was obtained from Promega (WI, USA). LIVE/DEAD Cell Imaging Kit was purchased from Molecular Probes, Life Technologies Corp. (CA, USA). Alizarin Red S, G418 disulfate salt, ascorbic acid, dexamethasone, β-glycerophosphate, and cetylpyridinium chloride were purchased from Sigma Aldrich (MO, USA).

**Fabrication of PCL/f-CNO nanocomposites**

The synthesis of PMPMA-CNOs was accomplished in good yield according to a previously reported protocol [27]. Two different concentrations of f-CNOs (0.2 wt% and 0.5 wt%) were utilized to fabricate PCL/f-CNO nanocomposites. For this, f-CNOs were probe-sonicated for 45 min in dichloromethane; then, PCL (45 grams) was added into the f-CNO solution, and probe sonication was continued for 120 min to accomplish a homogeneous black solution containing PCL-coated CNOs. After that, the black solution of PCL/f-CNOs was dried in a hot oven at 50 °C until it was completely dried. Then, the uniform black PCL/f-CNOs was loaded on 2 mm of stainless steel molds and placed in a CARVER 4122 hydraulic heating press at 120 °C followed by 24 MPa pressure for 15 min. After that, the pressure was released,
and molds were cooled to room temperature to obtain PCL/f-CNO nanocomposite sheets.

Characterizations of PCL/f-CNO nanocomposites

The morphological properties of synthesized PCL/f-CNOs were systematically characterized using scanning electron microscopy (SEM; ZEISS EVO® MA 25, Ostalbkreis, Baden-Württemberg, and Germany). PCL/f-CNO composite sheets were immersed in liquid nitrogen to fracture into small sections and gold-coated before measuring SEM images. The differential scanning calorimeter (DSC; TA Q2000, New Castle, DE) was used to measure the melting temperature and crystallinity of the nanocomposites at a scan rate of 10 °C/min with a temperature range of 25–120 °C under nitrogen atmosphere. The thermal gravimetric analysis (TGA; SDT Q600, TA Instruments) was used to measure the mass loss of the nanocomposites at a scan rate of 10 °C/min with a temperature range of 25–600 °C under nitrogen atmosphere. The mechanical properties of the nanocomposites were recorded using a tensile testing machine (Instron 3365, Instron, and Norwood, MA) according to ASTM D 638. The size of the tensile sample was 33.0 × 6.0 × 2.0 mm, and the crosshead speed was 50 mm/min. The roughness (Ra) of PCL/f-CNO nanocomposites was measured using the Mitutoyo profilometer (Mitutoyo, Guadalup, Nuevo Leon, Mexico).

Cytotoxicity evaluation of PCL/f-CNO nanocomposites

Human osteoblast cells (bone-forming cells) were utilized to evaluate the biocompatibility or in vitro cytotoxicity of PCL/f-CNO nanocomposites. The cytotoxicity of PCL/f-CNO nanocomposites was studied by cultivating osteoblast cells on the surface of PCL/f-CNO (0.2 wt%) and PCL/f-CNO (0.5 wt%) nanocomposite samples. Additionally, cell viability and cell morphology were measured.

Cell viability on PCL/f-CNO nanocomposites

The cell viability was measured using the CellTiter96® AQuous One Solution Cell Proliferation Assay. Initially, disk-shaped (~6.3 mm in diameter) specimens of pure PCL and PCL/f-CNO composites were prepared to measure the proliferation and cell viability. Prior to the cell seeding, the disk-shaped specimens were sterilized with ethanol (70% v/v) followed by UV irradiation. Then, the specimens were cultured on 96-well plates, and osteoblast cells at a density of 1 × 10^4 cells/mL were seeded on the surface of the samples in Dulbecco’s modified Eagle’s medium/F12 (DMEM/F12) supplemented with 2.5 mM of L-glutamine, 0.3 mg/mL of G418 disulfate salt, 10% of fetal bovine serum, and 1% of penicillin/streptomycin. The cells were nurtured for 24 h in a 5% CO₂ atmosphere. Then, the medium was discarded from the wells and rinsed several times with phosphate buffer solution. After 14 days, cell images were recorded by an optical microscope (Model IN200A-5M, Amscope, Chino, CA).

Cell attachment and morphology studies on PCL/f-CNO nanocomposites

The cell culture measurements were conducted to quantify the cell attachment and cell proliferation of nanocomposites. The morphological topographies of osteoblast cells on the surface of nanocomposites were measured by SEM. For this, osteoblast cell–cultivated nanocomposite specimens were nurtured for 14 days. Next, the samples were washed with PBS and cured with 2.5% glutaraldehyde for 12 h. After that, the specimens were dehydrated with ethanol and dried in a vacuum desiccator for 12 h. The dried nanocomposite specimens were gold-coated prior to SEM analysis.

Statistical analysis

All the quantitative data were expressed as mean ± standard deviations with n = 3. Statistical analysis was performed by one-way analysis of variance (ANOVA) followed by Tukey’s post hoc tests using Minitab17 (Minitab, State College, PA). The statistical significance was considered at P < 0.05.

Acknowledgments

We gratefully acknowledge Consejo Nacional de Ciencia y Tecnología de México and Tecnologico de Monterrey.

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