Here we address the question for the case of Fe\textsuperscript{1} the question how to get the magnetic structure of a necessary component for the spintronics [3], thus magnetic magnetic structure. providing that the electrons interact with the known chiral manipulate spin polarization of an electron current proving that the electrons interact with the known chiral magnetic structure.

The ability to manipulate the electron spin is a necessary component for the spintronics [3], thus magnetic chiral organic molecules [4] or large scale magnetic structures have been proposed as such tools [5]. However, the question how to get the magnetic structure of a necessary chirality for spintronics applications is still open. Here we address the question for the case of Fe\textsubscript{1−x}Co\textsubscript{x}Si solid solutions which, for certain compositions, show chiral (spiral) magnetic ordering [6–8].

The structural chirality in monosilicides of 3d-metals is solely controlled by crystal growth [9]. A link between the structural and magnetic chiralities is provided by the Dzialoshinskii-Moriya interaction (DMI) and has been experimentally proved for many monosilicides of 3d-metals [9,12]. The strength of the DMI defines the pitch of the magnetic spiral while the sign of the DMI sets a relationship between structural and magnetic chiralities to be the same or opposite [7,13].

For powder samples of Mn\textsubscript{1−x}Fe\textsubscript{x}Ge [12,14] and Fe\textsubscript{1−x}Co\textsubscript{x}Ge [15] it was shown that the spiral wave vector \( k = 2\pi/d \), where \( d \) is the spiral period, goes to zero value at a certain composition. The monotonic behavior of the wave vector indicates that the DMI goes to zero at the very same composition and, therefore, should change its sign as a function of \( x \) [12,15]. Recently, it was possible to reproduce the observed change of \( D \) in Mn\textsubscript{1−x}Fe\textsubscript{x}Ge using density-functional theory calculations [16,17]. The dynamic of the \( d_{xy}^{-} \)-like states has been revealed as the main mechanism behind the change of the sign of \( D \). With increasing \( x \) in Mn\textsubscript{1−x}Fe\textsubscript{x}Ge the \( d_{xy}^{-} \)-like states move from above to below the Fermi energy, become occupied and enter the region of \( d_{xy}^{-} \)-states with opposite spin, leading to the change of the sign of the DMI [17].

Here we further exploit the idea to control the DMI sign for the monosilicide series Fe\textsubscript{1−x}Co\textsubscript{x}Si. At variance with the germanides, the silicides can be grown as single crystals with controlled structural chirality [9]. A large size of crystals also makes possible a combined determination of the structural \( \Gamma_c \) and magnetic \( \gamma_m \) chiralities by resonant x-ray diffraction and polarized neutron scattering, correspondingly. Here and below chiralities are defined according to Ref. [13]. Taken together the two experimental probes allow us to follow \( \Gamma_c \times \gamma_m \) as a function of composition \( x \). Thus, the sign of the DMI term in the Hamiltonian of Ref. [7,8] can be experimentally probed via the product \( \Gamma_c \times \gamma_m \), allowing us to directly observe the flip of the link of the structural and magnetic chirality with the concentration \( x \).

Single crystals of Fe\textsubscript{1−x}Co\textsubscript{x}Si were grown using the Czochralski technique for the following concentrations \( x = 0.5, 0.6, 0.65, 0.7, 0.8 \). The same structural chirality of all grown crystals was provided by a consequent use of every grown crystal as the seed for the next one. As it was shown before this technique gives almost 100% control of the structural chirality [9]. The absolute crystal structure can be established by the X-ray single crystal diffraction data providing that resonant contribution enables to observe violation of the Friedel law, more details can be found in [18,20].
Single crystal Bragg diffraction data were collected at the room temperature using the PILATUS@SNBL diffractometer at the BM01A end station of the Swiss-Norwegian Beamlines at the ESRF (Grenoble, France); the wavelength of the synchrotron radiation was set to 0.70135 Å. The data were collected with a single θ-scan with angular step of 0.1° in a shutter-free mode with the Pilatus2M detector. The raw data were preprocessed with SNBL Toolbox, the integral intensities were extracted from the frames with the CrysAlisPro software [21], the crystal structure was solved with SHELXS and refined with SHELXL [22]. Crystals with an average size of about 100 microns were cut from large single crystals. The diffraction data summarized in Table I.

| x   | \( R_1 \)  | \( R_1 \)  | \( R_2 \)  | \( x_{\text{Fe}} \)  | \( x_{\text{Si}} \)  | Flack  |
|-----|---------|---------|---------|----------------|----------------|-------|
| 0.6 | 0.025   | 0.0295  | 0.0811  | 0.86000(2)     | 0.1579(5)      | 0.02(5) |
| 0.65| 0.017   | 0.0110  | 0.0290  | 0.85989(9)     | 0.1572(2)      | 0.01(4) |
| 0.7 | 0.016   | 0.0299  | 0.0665  | 0.85968(19)    | 0.1575(4)      | 0.10(7) |
| 0.8 | 0.049   | 0.0193  | 0.0420  | 0.85882(13)    | 0.1571(3)      | -0.01(7) |

The data are of good quality and agreement with the structural P2_13 model is high as can be seen from R-factors. The unit cell dimensions follow the Vegard law but the atomic positions stay nearly the same as a function of composition; the absolute structure is defined according to their values. Thus, in agreement with definitions given in Ref. [18, 23], the chirality \( \Gamma \) of structure with \( x_{\text{Fe}} \approx 0.86 \) is set to +1. The Flack parameter, which is a measure of presence of domains with the opposite chiral, is zero within 1÷2 standard deviations; the results confirm the same absolute structure (i.e. the same structures given in Refs. [18, 23], the chirality \( \Gamma \)) for all the tested crystals, as expected from the crystal growth procedure.

Fe_{1-x}Co_xSi compounds are magnetically ordered in the concentration range 0.05 \( \leq x \leq 0.8 \) [24, 25]. Magnetic measurements of newly synthesized samples were carried out with the SQUID-magnetometer Quantum Design MPMS-5S. Fig. 1 gives the temperature scans of the magnetization for different compounds at the field \( H = 100 \) mT. The experimental magnetization curves were used to estimate the ordering temperatures \( T_c \) as the position of the maximums at the derivative \( dM/dT \) (Fig.1).

The same analysis has been applied to the SQUID data for the samples studied in Ref. [18] for \( x = 0.1 \div 0.5 \). The \( x \)-dependence of the critical temperature \( T_c \) in the range \( x = 0.1 \div 0.7 \) is shown in Fig. 2. \( T_c \) increases monotonically on increase of \( x \) from 0.1 \( \div 0.4 \). For \( x > 0.4 \): \( T_c \) decreases again monotonically with \( x \) and approaching 0 at \( x \rightarrow 0.8 \), proving that the compounds under study are magnetically ordered up to \( x = 0.7 \). Notably, the pure compounds do not show any magnetic ordering, while their solid solutions show a remarkable compositional dependence of the ordering temperature. If the exchange interactions is a function of the number of Fe-Co pairs, then for an ideal mixture the maximum of \( T_c \) is expected at \( x = 0.5 \); the experiment gives \( x = 0.4 \) and the reason for the difference is still to be found.

The chirality of the magnetic structure was determined using polarized neutron diffraction [20, 27]. We used the protocol similar to one described in Refs. [12, 18] for the data analysis. The polarized small-angle neutron scattering (SANS) was carried out at the SANS-1 instrument at the Meier-Leibniz-Zentrum in Garching. The wavelength of the neutron beam was set in the range from 0.6 mm to 1.2 mm depending on the needed resolution. A position sensitive detector with 128 \( \times \) 128 pixels and a pixel size of 8 mm was used. These settings allowed us to cover a \( Q \) range from \( 2 \times 10^{-2} \) to 1 nm\(^{-1}\). The initial polarization of the neutron was \( P_0 \approx 0.9 \).

Figure 3 shows the polarized small angle neutron scattering maps for the compounds MnSi and Fe_{1-x}Co_xSi with \( x = 0.5, 0.6 \) and 0.7 at low temperature. As one can see, the MnSi reference sample shows a maximum of the scattering intensity at the right part of the de-
The helix wavevector $|k_s|$ has been extracted from the scattering maps at low temperature ($T \approx 3.5$ K). Figure 4 shows the $x$ dependence of $|k_s|$, the product of the lattice chirality $\Gamma_c$ and the magnetic chirality $\gamma_m$ is shown in Fig 4b. For $|k_s|$ the value increases from $|k_s| = 0.121$ nm$^{-1}$ for $x = 0.1$ to a maximum of $|k_s| = 0.185$ nm$^{-1}$ for $x = 0.2$. For $x > 0.2$ the value decreases to a minimum $|k_s| \rightarrow 0$ nm$^{-1}$ at the critical concentration of $x_c = 0.65$ and increases again to $|k_s| = 0.026$ nm$^{-1}$ for $x = 0.7$. The helix wavevector $k$ and the Dzyaloshinskii constant $D$ are linked via the equation:

$$k = \frac{SD}{A},$$

where $S$ is an average spin per unit cell and $A$ is the spin wave stiffness $[28]$. The spin wave stiffness and the spin value are expected to be monotonic functions of the Co content $[29, 30]$, therefore $|k_s| \rightarrow 0$ implies that $|D| \rightarrow 0$ at $x_c$.

The same concentration $x_c$ separates two regions with opposite values of the product $\Gamma_c \times \gamma_m$, therefore, $D$ not only goes through zero at $x_c$ but also changes its sign.

To summarize, we show that the chiral magneto-lattice coupling mapped phenomenologically as the DMI could be applied to control magnetic chirality as needed for yet illusive spintronics applications.
The sign of the Dzyaloshinskii constant $D$ defines the chirality of magnetic helix relative to the structural chirality. The product $\text{sgn}(D) \times \Gamma_c \times \gamma_m$ is an invariant with respect to inversion and time-reversal operations ensuring that left-handed and right-handed polymorphs have the same energy. The sign of $D$ depends on 3d-element occupying the metal site in $\text{Fe}_{1-x}\text{Co}_x\text{Si}$, and also in monogermanides \cite{12,14,15}. The difference in the critical concentrations, $x_c = 0.65$ for monosilicide and $x_c = 0.6$ for monogermanides \cite{15} is rather small; more detailed sampling near the critical concentration has to be done to find whether this difference is significant. Interesting to note that the different systems $\text{Mn}_{1-x}\text{Fe}_x\text{Ge}$ also shows similar behavior with a relatively close concentration $x_c = 0.75$. Those findings together with a complex nature of the transformation of the helical magnetic structure to a ferromagnetic-like at $x \rightarrow x_c$ should be subject of further theoretical and experimental studies.

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