Enhanced DSSCs efficiency via Cooperate co-absorbance (CdS QDs) and plasmonic core-shell nanoparticle (Ag@PVP)

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This paper describes cooperate the co-absorbance (CdS QDs) and the plasmonic core-shell nanoparticles (Ag@PVP) of dye synthesized solar cells in which CdS QDs and Ag@PVP are incorporated into the TiO2 layer. Cooperative nanoparticles show superior behavior on enhancing light absorption in comparison with reference cells. Cooperated DSSC exhibits the best performance with the power conversion efficiency of 7.64% which is superior to that of the free–modified DSSC with the PCE of 5%. Detailed studies offer an effective approach to enhance the efficiency of dye synthesized solar cells.

The dye-sensitized solar cells (DSSCs) have been regarded as a promising branch of study contrasted to conventional silicon-based photovoltaic devices for sustainable energy supply at low cost and high environmental friendliness1. Titanium dioxide (TiO2) which absorbs the sunlight in the ultraviolet region2 is one of the most prominent oxide materials for performing various kinds of industrial applications such as photovoltaic1, photocatalytic4,5. The wide band gap of TiO2 limits its absorption to the ultraviolet region of the solar spectrum6. In addition, the low electron mobility of TiO2 leads to inferior conversion efficiency of solar cells7,8. Until the present time, many assays have been done by focusing on the development of high performance sensitizers (e.g. different dyes) to enhance light harvesting in the visible light region9–12. While the development of new dyes has led to continued improvements in the efficiency of DSSCs in recent years13,14, they are still restricted by the weak absorption of the dye sensitizer like ruthenium(II) polypyridyl dyes (e.g. N719). N719 absorbs strongly at 535 nm, but it has drastically reduced extinction coefficients at longer wavelengths15. Therefore, enhancing the light harvesting efficiency (LHE) in the 300–900 nm wavelength range is considered as a favorable approach to increase the power conversion efficiency (PCE) and photocurrent of these devices16,17. It is well known that an effective method for trapping light or enhancing light harvesting to develop new photovoltaic devices is utilizing of some noble metallic nanostructures with high scattering cross section18,19. It has been reported that metal nanoparticles (NPs) such as gold (Au) and silver (Ag) can enhance the photo response of photovoltaic devices by acting as light trapping agents20,21, photosensitizers22, and electron traps for facilitating charge separation23,24. These nanoparticles have a strong optical behavior in the visible region due to surface plasmon resonance (SPR) effect due to collective electron oscillation25.

Also Quantum dot-sensitized solar cells are considered as a promising candidate for the progression of next generation solar cells due to their simple and low cost fabrication techniques. Some quantum dots (QDs) such as CdS, CdSe, CdTe, PbS and so on, which absorb light in the visible range, can serve as co-sensitizers so after absorption of a photon with enough energy, they are able to transfer electrons to the conduction band of TiO2.4. Among these QDs, CdS with suitable band gap and band positions compared to the conduction band of TiO2 can create a long distance charge separated states with electron and hole at sites far from each other, so it’s proper for using in these devices, in order to improve energy conversion efficiency of QDSSCs13,14. However, the relatively low conversion efficiency of such cells is still a primary challenge for the large-scale applications of QDSSCs. QDs have the advantage of a broad absorption spectra compared to molecular dyes with narrow absorption spectra.

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Moreover, photochemical reactions, particularly with the liquid electrolyte can induce significant degradation of the QD sensitizers. One can find detailed accounts related to these subjects in two recent reviews. In this work we have reported a cooperative PCE enhancement of 60% by adding CdS QDs and Ag@PVP nanoparticles mixture into TiO₂ layer.

We have presented a design which combines the benefits of QDs in terms of their broad absorption spectrum and the light harvesting efficiency (LHE) of noble metal (Ag@PVP). In this design, QDs show significant absorption between 200–300 nm, while N719 dye molecules show absorption in 313 nm and 500–560 nm, and Ag@PVP NPs have absorption in 350–450 nm (Fig. 1).

In addition, the effects of different weight volume percent (w/v) and treatment time of Ag@PVP nanoparticles (NPs) on performance of devices were studied.

**Result and Discussion**

Here, we have introduced a very simple and low cost method for synthesis of CdS QDs. In this method, CdS QDs were prepared by co-precipitation method in which mixture of water, propylene glycol, and ethanol were used as solvent in 40–50 °C. Figure 2a,b show the XRD pattern and HRTEM of as-synthesized CdS quantum dots where CdS QDs were prepared with no impurity and uniform size.

In order to investigate the effect of CdS QDs and Ag@PVP NPs in DSSCs, we first optimized device performance based on Ag@PVP NPs, and then we investigate their cooperative effect on DSCs performance a case by case. The devices were assembled with different weight percent of Ag@PVP from 0.33% to 1%. These devices were fabricated and measured under AM 1.5 illuminations at 100 mW/cm². Corresponding current density versus voltage (J–V) curve and details of DSCs based on the different weight percent of Ag@PVP NPs are reported in Fig. 3 and Table 1.

The best efficiency of cells which immersed in 0.33% of Ag@PVP for 2 min was 7.14% (average efficiency was 7.04%). It shows 43% improvement compared to reference cells (5%). 8.01% efficiency (average 7.9) was achieved by increasing Ag@PVP concentration into 0.5% for 2 min. When photoanodes immersed in 1% of Ag@PVP, efficiency decreased to 3.2%. At first, by increasing the concentration of plasmonic materials to 0.5%, efficiency increased sufficiently because more N719 molecules are effected by near field of plasmonic materials while high concentrations of Ag@PVP can act as trap center and decrease the efficiency. Therefore, efficiency decreased to 3.2 (average 3) in the present of 1% Ag@PVP.

According to our previous work, we found that treatment time has a sufficient effect on the efficiency. Thus, we optimized the treatment time for immersing photoanode in Ag@PVP solution. We investigated three
treatment time (1, 2, and 3 min). As seen in Fig. 3, optimum treatment time is 2 min. Treatment time has the same effect as concentration of Ag@PVP. The efficiency of the devices with 1 min treatment was 6% (average 5.9%) which shows better performance compared to reference cells and efficiency of cells with 2 min treatment was 7.14%.

Figure 2. (a) XRD pattern of as synthesized CdS QDs, (b) related HRTEM.

Figure 3. (a) J-V curve of different devices including different weight percent of Ag@PVP (b) Jsc & □% -weight percent of Ag@PVP and (c) Voc & FF -weight percent of Ag@PVP.
While increasing time treatment to 3 min showed the reverse effect on performance of DSSCs (Fig. 4 and Table 1).

After that, we studied the cooperation effect of CdS QDs with Ag@PVP NPs by simply mixing them in optimized conditions together into TiO₂ layer. Very interestingly, devices with CdS QDs/Ag NPs had much better solar cell performance.

We chose 0.33% w/v of Ag%PVP and CdS QDs to investigate their cooperation. Then we immersed the photoanodes in this solution for 2 min. Typically, solar cells with CdS QDs/Ag@PVP NPs showed 6.21% efficiency (average 6.19%) which is lower than 7.14% for solar cells with Ag@PVP NPs. So we increase treatment time to 3 min led to improve efficiency to 7.65% which shows 7% improvement compared to 0.33% of Ag@PVP (Fig. 5 and Table 1). As shown in Fig. 5 and Table 1, Jsc was reached to 22.5 by cooperation of Ag@PVP and CdS QDs.

Figure 6 shows the DRS spectra of reference cell and cell including CdS QDs/Ag/PVP. According to this Figure, cell including CdS QDs/Ag/PVP shows broad absorption from 200–600 nm, which peak in 350–450 nm refers to the plasmonic peak of Ag@PVP.

The small standard deviations indicate good reliability and reproducibility of our results. Intensity modulated photocurrent spectroscopy IMPS²⁹,³⁰ and intensity modulated photovoltage spectroscopy IMVS⁸ can be used to measure the respective time constants for the combined processes of charge collection and charge recombination at short circuit and charge recombination at open circuit condition in dye-sensitized solar cells.

It is expected that the loss of injected electrons via both photocurrent from external load and recombination are proportional to the electron concentration in TiO₂ and the rate constants for these processes are independent.

| DSCs         | Voc   | Jsc   | FF   | η%  |
|--------------|-------|-------|------|-----|
| 0.33% Ag@PVP | 0.69 ± 0.01 | 20.73 ± 0.15 | 0.50 ± 0.02 | 7.14 |
| 0.5% Ag@PVP  | 0.67 ± 0.01 | 22.16 ± 0.20 | 0.54 ± 0.02 | 8.01 |
| 1% Ag@PVP    | 0.69 ± 0.02 | 15.25 ± 0.25 | 0.30 ± 0.01 | 3.2  |
| Ref.         | 0.71 ± 0.03 | 12.57 ± 0.20 | 0.52 ± 0.01 | 5    |
| 1 min        | 0.65 ± 0.03 | 14.43 ± 0.1  | 0.66 | 6.21 |
| 2 min        | 0.69 ± 0.02 | 20.73 ± 0.1  | 0.50 | 7.14 |
| 3 min        | 0.61 ± 0.03 | 10.04 ± 0.2  | 0.64 | 4.01 |
| 0.33% Ag@PVP (2 min) | 0.69 ± 0.02 | 20.73 ± 0.1 | 0.50 ± 0.02 | 7.14 |
| 0.33% Ag + 0.33% QD (2 min) | 0.65 ± 0.01 | 14.43 ± 0.1 | 0.66 ± 0.01 | 6.21 |
| 0.33% Ag + 0.33% QD (3 min) | 0.71 ± 0.03 | 21.25 ± 0.2 | 0.5 ± 0.01 | 7.65 |

Table 1. Solar cell performance parameters of fabricated devices under AM 1.5.

Figure 4. (a) Voc & FF – time treatment of Ag@PVP and (c) Jsc & η% - time treatment of Ag@PVP.
of each other. The time constant for charge recombination at open circuit ($\tau_r$) was obtained from IMVS. The collection (transport) time ($\tau_c$) can be estimated from the equation $\tau_r = 1/2\pi f_c$, where $f_c$ is the characteristic frequency maximum of the IMPS imaginary component and the charge-collection efficiency $\eta_{cc}$ is given by the equation $\eta_{cc} = 1 - (\tau_c/\tau_r)$. IMPS and IMVS experiments were done by a modulated AC light from 1 to 1000 Hz superimposed on a relatively large bias illumination at 470 nm.

The effect of CdS QDs, and Ag@PVP NPs on transport, recombination times, and charge-collection efficiency are presented in Table 2. It is clear that devices based on CdS QDs and Ag@PVP NPs show higher collection time and charge-collection efficiency. IPCE was used for explaining the improved PCEs. Figure 7 shows the IPCE spectra of control device with best efficiency and devices including CdS QDs and Ag@PVP. It is indicated that the IPCE of devices including CdS QDs and Ag@PVP is increased over the whole wavelength range, especially within 400–700 nm, compared with that of TiO2/N719-only DSCs. The IPCE is related with the light-harvesting efficiency, electron injection efficiency, and carrier collection efficiency simultaneously. Assuming the electron...
injection and collection are seldom affected, the increase of the IPCE should be attributed to the light harvest enhancement due to the CdS QDs and Ag@PVP.

Based on these results, we illustrate a mechanism for this improvement (Fig. 8). As shown in mechanism, PCE could improve in 2 ways. First, CdS QDs can act as co-sensitizer. It means CdS absorb certain wavelengths and produce electrons and holes which can transfer to the external circuit. Also conduction band of CdS is located between the conduction band of dye and TiO2. Therefore electrons can jump from the conduction band of dye to the conduction band of CdS and finally to the conduction band of TiO2 before there are recombined with holes.

Table 2. The effect of Ag@PVP, and Ag@PVP/CdS QDs on transport, recombination times, and charge-collection efficiency.

| DSCs                        | τc(s)          | τr(s)          | ηcc         |
|-----------------------------|----------------|----------------|-------------|
| Ref                         | 1.00E-4        | 6.02E-4        | 0.81        |
| 0.33 Ag@PVP (2 min)         | 0.012E-4       | 0.43E-4        | 0.97        |
| Ag@PVP/CdS QDs (2 min)      | 0.825E-4       | 0.0082E-4      | 0.99        |
| Ag@PVP/CdS QDs (3 min)      | 0.0088E-4      | 0.894E-4       | 0.99        |

Figure 7. IPCE spectra of control device with best efficiency and devices including CdS QDs and Ag@PVP.

Figure 8. Detailed mechanism for enhancing efficiency in CdS QdS/Ag@PVP cell.
in valance band of dye (Fig. 8a). Second, the local plasmonic resonances arising from Ag@PVP core shell nanoparticles increase the optical absorption of dye molecules in DSCs. PVP has the advantages of not only readily forming films, but also can chemically and electronically protect the silver from the corrosion and recombination. Besides, PVP have the unique ability of adhesion to dye molecules, which help to trap mass of dye molecules surrounding the silver nanoparticles surface, hence the LSP field from silver sphere can affect sufficient dye molecules, as shown in Fig. 8b, resulting in a significant enhancement on the optical absorption and overall efficiency of the DSCs.

Figure 9 shows detailed evaluation of the device parameters during the 400 hours measured at room temperature under A.M 1.5. During this time period, PCE decreased 53%. During the first 50 hours PCE decreased suddenly because iodide/triiodide electrolyte corrodes CdS QDs. After 50 hours, this rate (rate of decrease in PCE) decreased and it was stopped after 300 hours.

Figure 10 and Table 3 demonstrate typical Nyquist plots of the DSSCs with bare TiO2 and cells containing CdS QDs/Ag@PVP. The impedance spectra were fitted to a simplified version of the equivalent circuit (Fig. 10). The

| DSCs            | τ (s) | K_{eff} (S^{-1}) | R_{ct} (Ω cm^2) | R_{ct} (Ω cm^2) | Deff (cm^2 s^{-1}) |
|-----------------|-------|------------------|-----------------|-----------------|-------------------|
| CdS QDs         | 0.075 | 13.33            | 45              | 120             | 3.55E-5           |
| Ref.            | 0.041 | 23.81            | 14              | 73              | 12.7E-5           |
| CdS QDs/Ag@PVP  | 0.043 | 23.25            | 28              | 110             | 9.13E-5           |

Table 3. Electrochemical parameters obtained by fitting the impedance spectra of DSSCs.
Figure 11. HRTEM of (a) as-synthesized Ag@PVP, (b) pure TiO2 film and (c) TiO2 film, including Ag@PVP and CdS quantum dots.
some important parameters of the working electrodes in these devices were determined from the second semi-
circles in Fig. 10 that the results have been summarized in Table 3. The effective diffusion coefficient of electrons
(D_{eff}) can be determined by Eq. 1:

\[ D_{eff} = \left( \frac{RCE}{RCE} \right) \frac{L^2 K_{eff}}{\tau} \]

(1)

Here, keff is the reaction rate constant for the electron recombination with triiodide and L is film thickness.
keff is obtained by the electron lifetime (\(\tau\)) and angular frequency (\(\omega_{max}\)) at the second semicircles in the Nyquist
plots of the dye synthesized solar cells (Eqs (2) and (3)):

\[ \tau = \frac{1}{\omega_{max}} \]

(2)

\[ K_{eff} = \frac{1}{\tau} \]

(3)

There is small difference between these devices for the RCE at the counter electrode/electrolyte interface.
These results show that the CdS QDs/Ag@PVP core shell NPs do not make the potential over the counter elec-
trode. Notably, the fitted value of R_{CT} for cells containing CdS QDs/Ag@PVP NPs are found ~110 \(\Omega\) cm², while
the corresponding value for reference cells are ~73 \(\Omega\) cm².

This significant increase in R_{CT} shows that the organic shells are more favorable to suppress the charge
recombination process that arises from electrons in TiO₂ film with triiodide in electrolyte solution and leads to
increase J_{sc}. On the other hand, the adsorption of CdS QDs/Ag@PVP NP on the TiO₂ surface causes to increase
electron density in the conduction band of TiO₂. In the cell based on CdS QDs/Ag@PVP NP electrode, the middle-frequency peak in the Nyquist plot shifts to lower frequency relative to reference device, exhibiting a
longer electron lifetime for these cells.

Conclusion
To conclude, we improved the PCEs and charge-collection efficiency of DSCs by incorporating QDs and Ag@
PVP NPs into TiO₂ layer. LSPR induced near field enhancement not only leads to heightened light absorption
of active layer materials of DSCs but also benefits charge separation and transport, resulting in increased charge
carrier density. Moreover, we also showed that cooperative plasmonic enhancement could be achieved by simply
combining CdS QDs and Ag@PVP NPs into TiO₂ layer. The LSPR and QDs absorption lead to enhanced light
absorption region after combination; this idea is further proved by solar cell performance of devices with separate
CdS, Ag@PVP NPs. We believe that the results of our study offer an effective approach to enhance the efficiency
of dye synthesized solar cells.

Method
Synthesis of CdS QDs. CdS QDs were prepared according to our previous work. Briefly, CdS QDs were
prepared by the reaction of cadmium oxalate with thioacetamide (TAA) in mixture of DI water, ethanol and propy-
lene glycol (PG) as solvent under magnetic stirring at 45 °C. Here, CdS QDs were prepared with a very simple
method at 45 °C.

To fabricate Ag@PVP core-shell NPs, monodisperse Ag@PVP NPs are prepared according to the method
described in our previous work. Briefly, an aqueous solution containing AgNO₃ (purchased from Merck) is
heated to its boiling point under stirring, and then a hydrazine solution is injected quickly into the system. An
aqueous solution of PVP (purchased from Merck) is added to the colloidal Ag solution to modify the Ag NPs. The
solution is stirred for 12–18 h at room temperature. The PVP-coating Ag NPs are collected by centrifugation and
re-dispersed in ethanol by bath sonication.

Preparation of working electrode. Electrophoretic deposition (EPD) method was utilized to prepare
P25 NPs-based films used in DSSCs. During Electrophoretic deposition, the cleaned FTO glass was remained at
a positive potential, anode, while a pure steel mesh was used as the cathode electrode. Power was supplied by a
Megatek Pro-grammable DC Power Supply (MP-3005D). During the process the applied voltage was 10 V and the
deposition cycles were 4 with each cycle of 15 s. The coated substrates were dried at 150 °C. The resulted layer
was annealed under air flow at 500 °C for 60 min. Electrolyte solution and the amount of additives are important
for creation of a high quality surface as mentioned in other experiment reported previously. We used optimal
concentrations of additives in the electrolyte solution as follows: I₂, 6 mg, acetone 0.4 ml, and water 1 ml. The elec-
trodes, which were prepared with EPD, were immersed in colloidal solution of Ag@PVP NPs and CdS QDs for
2 min before drying at 100 °C. Figure 11 shows the HRTEM of as-synthesized Ag@PVP (Fig. 11a), working elec-
trode before (Fig. 11b) and after immersing in colloidal solution of Ag@PVP NPs and CdS QDs (Fig. 11c). As seen
from Fig. 11a, size of Ag core and PVP shell are about 20 nm and 5 nm, respectively. Ag@PVP is almost spherical.
Figure 11b shows the pure P25 particles with size about 21 nm. HTRTEM of Ag@PVP NPs, CdS QDs and P25
blend is shown in Fig. 11c. As seen in the figure, Ag@PVP NPs and CdS QDs are well dispersed in P25 particles.

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Acknowledgements
This work is financially supported by University Malaya Research Grant (RP038B-15HTM) and the council of University of Kashan by Grant No (159271/779).

Author Contributions
O.A. did the experiments, interpreted the results and wrote the main manuscript text, S.B., A.T.Y. and M.S.-N. helped in characterization. All authors reviewed the manuscript.

Additional Information

Competing financial interests: The authors declare no competing financial interests.

How to cite this article: Amiri, O. et al. Enhanced DSSCs efficiency via Cooperate co-absorbance (CdS QDs) and plasmonic core-shell nanoparticle (Ag@PVP). Sci. Rep. 6, 25227; doi: 10.1038/srep25227 (2016).