Improved efficiency of NiO$_X$-based p-i-n perovskite solar cells by using PTEG-1 as electron transport layer

Bart G. H. M. Groeneveld, Mehrdad Najafi, Bauke Steensma, Sampson Adjokatse, Hong-Hua Fang, Fatemeh Jahani, Li Qiu, Gert H. ten Brink, Jan C. Hummelen, and Maria Antonietta Loi

Citation: APL Materials 5, 076103 (2017); doi: 10.1063/1.4992783
View online: http://dx.doi.org/10.1063/1.4992783
View Table of Contents: http://aip.scitation.org/toc/apm/5/7
Published by the American Institute of Physics

Articles you may be interested in

The effect of hole transporting layer in charge accumulation properties of p-i-n perovskite solar cells
APL Materials 5, 076102 (2017); 10.1063/1.4991030

Chlorinated fluorine doped tin oxide electrodes with high work function for highly efficient planar perovskite solar cells
Applied Physics Letters 110, 263901 (2017); 10.1063/1.4989560

Low-cost electrodes for stable perovskite solar cells
Applied Physics Letters 110, 233902 (2017); 10.1063/1.4984284

Tuning of perovskite solar cell performance via low-temperature brookite scaffolds surface modifications
APL Materials 5, 016103 (2017); 10.1063/1.4973892

Direct observation of dramatically enhanced hole formation in a perovskite-solar-cell material spiro-OMeTAD by Li-TFSI doping
Applied Physics Letters 110, 123904 (2017); 10.1063/1.4977789

Electric and photovoltaic characteristics of a multi-layer ReS$_2$/ReSe$_2$ heterostructure
APL Materials 5, 076101 (2017); 10.1063/1.4991028
We present efficient p-i-n type perovskite solar cells using NiO$_x$ as the hole transport layer and a fulleropyrrolidine with a triethylene glycol monoethyl ether side chain (PTEG-1) as electron transport layer. This electron transport layer leads to higher power conversion efficiencies compared to perovskite solar cells with PCBM (phenyl-C$_{61}$-butyric acid methyl ester). The improved performance of PTEG-1 devices is attributed to the reduced trap-assisted recombination and improved charge extraction in these solar cells, as determined by light intensity dependence and photoluminescence measurements. Through optimization of the hole and electron transport layers, the power conversion efficiency of the NiO$_x$/perovskite/PTEG-1 solar cells was increased up to 16.1%. © 2017 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses/by/4.0/).

Hybrid inorganic-organic perovskite solar cells have attracted considerable attention because of their high power conversion efficiencies, owing to their strong absorption over a wide range of the solar spectrum, long electron–hole diffusion lengths, and ambipolar charge transport characteristics. These solar cells have been reported both in the n–i–p and in the p–i–n structure. However, n–i–p type planar cells, where the perovskite layer is sandwiched between TiO$_2$ and the hole transport layer, suffer from severe hysteresis, as opposed to p–i–n cells. A commonly used hole transport layer (HTL) in p-i-n solar cells is poly(3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS). However, perovskite solar cells with this HTL typically have a lower open-circuit voltage ($V_{OC}$) than the cells based on TiO$_2$. In addition, solar cells incorporating PEDOT:PSS are sensitive to degradation, which is detrimental for long-term operation.

Inorganic materials, such as metal oxides, are more stable alternatives as HTLs because they are less prone to degradation under a variety of conditions. NiO$_x$ is an example of an inorganic HTL with good chemical stability and suitable work function (~4.8 eV) for p-i-n perovskite solar cells. The $V_{OC}$ of NiO$_x$-based perovskite solar cells is typically higher than that of PEDOT:PSS devices (1 V and 0.9 V, respectively), leading to higher power conversion efficiencies.

PCBM (phenyl-C$_{61}$-butyric acid methyl ester) is commonly used as the electron transport layer in the p-i-n structure, but this material is often combined with other layers, such as batohcuprine (BCP), to block holes and improve the power conversion efficiency. Incorporating additional layers into the solar cell’s device structure leads to increased complexity in processing, and therefore it is desirable to replace the electron transport layer to simplify the overall device structure. Recently, Yin et al. showed NiO$_x$-based solar cells with only PCBM as the electron transport layer (ETL), reaching a power conversion efficiency (PCE) of 15.7%. However, their solar cells have a $V_{OC}$ lower than the cells based on TiO$_2$. In addition, solar cells incorporating PEDOT:PSS are sensitive to degradation, which is detrimental for long-term operation.

B. G. H. M. Groeneveld and M. Najafi contributed equally to this work.

Author to whom correspondence should be addressed: m.a.loi@rug.nl
than 1 V, which limits their efficiency. Our previous work on PEDOT:PSS-based p-i-n perovskite solar cells demonstrated that the PCE increases when PCBM is replaced by PTEG-1, which is a fulleropyrrolidone with a triethylene glycol monoethyl ether side chain.\textsuperscript{9,20}

In this paper, we report the first use of PTEG-1 in perovskite solar cells with a NiO\textsubscript{x} hole transport layer. We compare the performance of solar cells with PTEG-1 to PCBM-based devices and conclude that the use of PTEG-1 can lead to higher PCEs up to 16.1\% with open-circuit voltages as high as 1.12 V.

The device structure used in this work is ITO/NiO\textsubscript{x}/CH\textsubscript{3}NH\textsubscript{3}PbI\textsubscript{3}/ETL/Al, as shown in Fig. 1(a). The chemical structures of the two materials used as ETLs, PCBM, and PTEG-1, are displayed in Fig. 1(b). The high quality and low roughness of the NiO\textsubscript{x} layer (Fig. S1 of the supplementary material shows the atomic force microscopy (AFM) image of a NiO\textsubscript{x} film) enables a good coverage of the perovskite film.

The perovskite layer is deposited from a mixture of high boiling point solvents, based on the method reported by Jeon \textit{et al.}\textsuperscript{6} Figure S2 of the supplementary material displays the scanning electron microscopy (SEM) images of a uniform and pinhole-free perovskite film deposited on a NiO\textsubscript{x} film. From these images it is clear that the perovskite grains are extremely large (on the order of tens to hundreds of microns) and that the surface is completely covered.

Figure 2(a) shows the current density–voltage ($J$–$V$) characteristics of the solar cells with PTEG-1 and PCBM as electron transport layers under AM1.5G illumination. The $J$–$V$ parameters are summarized in Table I. The solar cell with PCBM reaches 11.8\% power conversion efficiency, with a short-circuit current density ($J_{SC}$) of 19.6 mA/cm\textsuperscript{2}, an open-circuit voltage of 1.09 V and a fill factor (FF) of 55\%. When PTEG-1 is used instead of PCBM, the $V_{OC}$ and $J_{SC}$ increase to 1.11 V and 21.4 mA/cm\textsuperscript{2}, respectively. This results in a higher PCE of 13.1\%, since the fill factor does not change.
TABLE I. Device performances for different ETLs.

| ETL    | $J_{SC}$ (mA/cm$^2$) | $V_{OC}$ (V) | FF (%) | PCE (%) |
|--------|----------------------|--------------|--------|---------|
| PCBM   | 19.6                 | 1.09         | 55     | 11.8    |
| PTEG-1 | 21.4                 | 1.11         | 55     | 13.1    |

Our group has previously reported improvements in $V_{OC}$ and $J_{SC}$ after replacing PCBM by PTEG-1 in hybrid perovskite devices using PEDOT:PSS as the hole extraction layer. This improvement was attributed to the reduction of trap-assisted recombination in the case of PTEG-1.\textsuperscript{9} We carried out light intensity dependence measurements to determine the reason for the increased performance in the current device structure and the results are displayed in Fig. S3 of the supplementary material. The dependence of $V_{OC}$ on the light intensity should be linear when plotted in a semi-logarithmic scale, which is the case in Fig. S3. We can extract the diode ideality factor $n$ (in units of $kT/q$, where $k$ is Boltzmann’s constant, $T$ is the absolute temperature, and $q$ is the elementary charge) from the slope of the fit. When $n$ is equal to 1, only bimolecular recombination takes place in the device. Trap-assisted recombination is characterized by $n$ between 1 and 2.\textsuperscript{21} In our case, the values of $n$ for PCBM and PTEG-1 are 1.37 and 1.21, respectively. This indicates that the trap-assisted recombination is lower in the case of PTEG-1. Figure 2(b) shows the external quantum efficiency (EQE) spectra of the ITO/NiO$_x$/perovskite/ETL/Al devices for both PCBM and PTEG-1. The EQE of the PTEG-1 device is higher than 80% over almost the entire wavelength range from 420 to 660 nm, and it is higher than that of PCBM over the full range of the measurement. The $J_{SC}$ obtained by integration of the EQE spectra is in agreement with the $J_{SC}$ obtained from the $J$–$V$ measurements for both ETL layers (see Fig. S4 of the supplementary material).

To fully understand why our PTEG-1 solar cells perform better than the PCBM-based devices, we conducted photoluminescence (PL) measurements (see Fig. 3). With these measurements it is possible to study the charge carrier recombination upon excitation. The steady state PL was measured for stacks composed of glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$/PCBM and glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$/PTEG-1; the structure glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$ was used as comparison. From the measurements in Fig. 3(a), it is clear that there is successful charge transfer from the perovskite layer to the ETLs, as indicated by the quenched PL signal. The quenching in the PTEG-1 sample is stronger than in the PCBM sample, which is supported by the time-resolved PL data shown in Fig. 3(b). This plot shows that the PL signal decays mono-exponentially for all samples. The lifetime of the PL signal is shorter for PTEG-1 ($\tau = 10.3$ ns) than for PCBM ($\tau = 25$ ns) and for the perovskite only ($\tau = 190$ ns). These results indicate that the charge extraction is better in the case of NiO$_x$/perovskite/PTEG-1 than in the case where PCBM is the ETL.

In order to improve the power conversion efficiency of the PTEG-1 solar cells, we attempted to optimize the thicknesses of the transport layers. Solar cells with different thicknesses of the PTEG-1

![FIG. 3. (a) Steady state PL spectra for glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$ (black line), glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$/PCBM (red line) and glass/NiO$_x$/CH$_3$NH$_3$PbI$_3$/PTEG-1 (blue line). (b) Time-resolved PL of the samples in (a). Symbols show the data, the solid lines are the mono-exponential fits.](image-url)
layer were made and the corresponding $J$-$V$ measurements are presented in Fig. 4(a) and in Table II. It was found that when the PTEG-1 layer is thin (about 35 nm), the devices show current leakage. As a result the fill factor is low, which could be due to insufficient coverage of PTEG-1 on the perovskite surface. Increasing the PTEG-1 thickness improves the FF and PCE considerably. The device with 50 nm PTEG-1 has a fill factor of 61%, but increasing the thickness further (60 nm) lowers the FF to 56%. Therefore, 50 nm is the optimal thickness for this ETL. Below this thickness, there is the possibility of shunt pathways. However, above 50 nm there will be additional series resistance in the layer, which reduces the PCE.

Figure 4(b) shows the effect of the NiO$_x$ thickness on the device performance; a slight change in the thickness has a large influence on the device performance. The thickness of the perovskite layer was 270 nm, the same as used in the cells from Fig. 4(a), and the PTEG-1 layers were 50 nm thick. The current density—voltage parameters are listed in Table III. It is evident that thicker layers lead to an increase in $V_{OC}$ (from 0.99 to 1.14 V). The short-circuit current density and fill factor for the device with 20 nm NiO$_x$ are 20.2 mA/cm$^2$ and 47%, respectively. When the thickness of the NiO$_x$ layer is increased to 40 nm, both $J_{SC}$ and FF increase. However, increasing the thickness beyond 40 nm does not lead to an overall better device performance, which is mainly caused by a reduction in the fill factor. This trend can be explained by the increased series resistance in the HTL. We can conclude here that the optimal thickness for NiO$_x$ is 40 nm, and this gives a solar cell with a $J_{SC}$ of 22.9 mA/cm$^2$, a $V_{OC}$ of 1.12 V, a FF of 63% and a resulting PCE of 16.1%. This is among the highest reported values for a NiO$_x$/perovskite/fullerene solar cell. The value of 16.1% is the maximum value

| PTEG-1 thickness (nm) | $J_{SC}$ (mA/cm$^2$) | $V_{OC}$ (V) | FF (%) | PCE (%) |
|-----------------------|----------------------|-------------|--------|---------|
| 35                    | 21.7                 | 1.10        | 39     | 9.4     |
| 50                    | 21.3                 | 1.11        | 61     | 14.4    |
| 60                    | 21.9                 | 1.10        | 56     | 13.6    |

| NiO$_x$ thickness (nm) | $J_{SC}$ (mA/cm$^2$) | $V_{OC}$ (V) | FF (%) | PCE (%) |
|------------------------|----------------------|-------------|--------|---------|
| 20                     | 20.2                 | 0.99        | 47     | 9.4     |
| 40                     | 22.9                 | 1.12        | 63     | 16.1    |
| 60                     | 20.9                 | 1.13        | 58     | 13.6    |
| 80                     | 21.1                 | 1.14        | 52     | 12.5    |
that we achieved; see Figs. S5 and S6 of the supplementary material for an overview of the power conversion efficiencies for devices prepared in the same way and the hysteresis in the champion device, respectively.

In summary, we have demonstrated that the power conversion efficiency of NiO$_x$-based p-i-n planar perovskite solar cells can be improved by changing the electron transport layer from PCBM to PTEG-1. This is mainly caused by the increased short-circuit current density when PTEG-1 is used. External quantum efficiency and photoluminescence measurements indicate that the charge extraction is more efficient for devices with PTEG-1 as ETL. In addition, we found that the solar cells with PTEG-1 suffer less from trap-assisted recombination compared to the PCBM counterparts. Optimization of the thicknesses of both the electron and hole transport layer lead to power conversion efficiencies up to 16.1% for ITO/NiO$_x$/CH$_3$NH$_3$PbI$_3$/PTEG-1/Al.

See supplementary material for details on device fabrication and characterization.

The authors are thankful to A. Kamp and T. Zaharia for technical support and to E. L. Ratcliff for discussions on nickel oxide layers. This work is part of the research programme of the Foundation for Fundamental Research on Matter (FOM), which is part of the Netherlands Organisation for Scientific Research (NWO). This is a publication of the FOM-focus Group “Next Generation Organic Photovoltaics,” participating in the Dutch Institute for Fundamental Energy Research (DIFFER).

1 M. M. Lee, J. Teuscher, T. Miyasaka, T. N. Murakami, and H. J. Snaith, Science 338, 643 (2012).
2 J. Burschka, N. Pellet, S.-J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin, and M. Grätzel, Nature 499, 316 (2013).
3 M. Liu, M. B. Johnston, and H. J. Snaith, Nature 501, 395 (2013).
4 H. Zhou, Q. Chen, G. Li, S. Luo, T.-h. Song, H.-S. Duan, Z. Hong, J. You, Y. Liu, and Y. Yang, Science 345, 542 (2014).
5 N. J. Jeon, J. H. Noh, W. S. Yang, Y. C. Kim, S. Ryu, J. Seo, and S. I. Seok, Nature 517, 476 (2015).
6 N. J. Jeon, J. H. Noh, Y. C. Kim, W. S. Yang, S. Ryu, and S. I. Seok, Nat. Mater. 13, 897 (2014).
7 J. H. Heo, H. J. Han, D. Kim, T. K. Ahn, and S. H. Im, Energy Environ. Sci. 8, 1602 (2015).
8 H. S. Kim, I. H. Jang, N. Ahn, M. Choi, A. Guerrero, J. Bisquert, and N. G. Park, J. Phys. Chem. Lett. 6, 4633 (2015).
9 S. Shao, M. Abdu-Aguye, L. Qiu, L.-H. Lai, J. Lii, S. Adjokatse, F. Jahani, M. E. Kamminga, G. H. ten Brink, T. T. M. Palstra, B. J. Kooi, J. C. Hummelen, and M. A. Loi, Energy Environ. Sci. 9, 2444 (2016).
10 M. P. de Jong, L. J. van IJzendoorn, and M. J. A. de Voigt, Appl. Phys. Lett. 77, 2255 (2000).
11 R. Betancur, M. Maymó, X. Elias, L. T. Vuong, and J. Martorell, Sol. Energy Mater. Sol. Cells 95, 735 (2011).
12 J. R. Manders, S.-W. Tsang, M. J. Hartel, T.-H. Lai, S. Chen, C. M. Amb, J. R. Reynolds, and F. So, Adv. Funct. Mater. 23, 2993 (2013).
13 Y. Sun, C. J. Takacs, X. Sun, R. Cowan, J. H. Seo, X. Gong, A. Roy, and A. J. Heeger, Adv. Mater. 23, 2226 (2011).
14 K. X. Steirer, P. F. Ndione, N. E. Widjonarko, M. T. Lloyd, J. Meyer, E. L. Ratcliff, A. Kahn, N. R. Armstrong, C. J. Curtis, D. S. Ginley, J. J. Berry, and D. C. Olson, Adv. Energy Mater. 1, 813 (2011).
15 J. Y. Jeng, K. C. Chen, T. Y. Chiang, P. Y. Lin, T. Da Tsai, Y. C. Chang, T. F. Guo, P. Chen, T. C. Wen, and Y. J. Hsu, Adv. Mater. 26, 4107 (2014).
16 J. You, L. Meng, T.-B. Song, T.-F. Guo, Y. (Michael) Yang, W.-H. Chang, Z. Hong, H. Chen, H. Zhou, Q. Chen, Y. Liu, N. De Marco, and Y. Yang, Nat. Nanotechnol. 11, 75 (2015).
17 J. Cui, F. Feng, H. Zhang, K. Cao, H. Yuan, Y. Cheng, F. Huang, and M. Wang, ACS Appl. Mater. Interfaces 6, 22862 (2014).
18 Z. Liu, A. Zhu, F. Cai, L. Tao, Y. Zhou, Z. Zhao, Q. Chen, Y.-B. Cheng, and H. Zhou, J. Mater. Chem. A 5, 6597 (2017).
19 X. Yin, Z. Yao, Q. Luo, X. Dai, Y. Zhou, Y. Zhang, Y. Zhou, S. Luo, J. Li, N. Wang, and H. Lin, ACS Appl. Mater. Interfaces 9, 2439 (2017).
20 F. Jahani, S. Torabie, R. C. Chiechi, L. J. A. Koster, and J. C. Hummelen, Chem. Commun. 50, 10645 (2014).
21 G.-J. A. H. Wetzelaer, M. Scheepers, A. M. Sempere, C. Momblona, J. Ávila, and H. J. Bolink, Adv. Mater. 27, 1837 (2015).