Preliminary Microstructural and Microscratch Results of Ni-Cr-Fe and Cr$_3$C$_2$-NiCr Coatings on Magnesium Substrate

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Abstract. Thermal coatings have a large scale application in aerospace and automotive field, as barriers improving wear mechanical characteristics and corrosion resistance. In present research, there have been used two types of coatings, Ni-Cr-Fe, respectively Cr$_3$C$_2$-NiCr which were deposited on magnesium based alloys (pure magnesium and Mg-30Y master alloy). There have been investigated the microstructural aspects through scanning electronic microscopy and XRD analysis and also a series of mechanical characteristics through microscratch and indentation determinations. The results revealed the formation of some adherent layers resistant to the penetration of the metallic indenter, the coatings did not suffer major damages. Microstructural analysis highlighted the formation of Cr$_3$C$_2$, Cr$_7$C$_3$, Cr$_3$Ni$_2$, Cr$_7$Ni$_3$, FeNi$_3$, Cr-Ni phases. Also, the apparent coefficient of friction for Ni-Cr-Fe coatings presents superior values than Cr$_3$C$_2$-NiCr coatings.

1. Introduction
Thermal coatings represent some special coatings processes, which has seen an accelerated development in the last decades, especially in automotive, naval, aerospace domains, like: turbine blades, cylinders, pistons, valves, in order to improve wear resistance, corrosion resistance and lifetime of the machine part [1–3]. The most common thermal deposition techniques are APS and HVOF, but there are also other coatings methods like: flame spraying, wire arc spraying, PLD etc [4–9]. Coatings obtained by plasma jet deposition (APS) with formation of Ni, Cr ceramic layers are among used materials in mechanical applications [10]. The main disadvantage of these coatings could represent the isolated inhomogeneity, micro-cracks and residual stresses. These stresses occur during cooling process of thermal spraying [11].

Mayrhofer et al. highlighted the wear resistance improvement of steel substrates by using Cr$_3$C$_2$ stabilized with NiCr layers in comparison with WC coatings [12]. It has been also shown that Cr$_3$C$_2$-NiCr coatings present comparable wear with tungsten carbide, but it has superior oxidation resistance [2]. Varies et al. studied the fatigue resistance improvement Cr$_3$C$_2$ based coatings by performing some kinetic energy layers and special treatments [13]. The researchers showed that the coatings which contain iron, improve the layers ability to make diffusion with base material being also cheaper and less pollutant with the environment than Co and Ni based coatings [14,15].

In the present paper, authors deposited two types of plasma jet coatings (Cr$_3$C$_2$-NiCr and Ni-Cr-Fe) on magnesium alloys substrate (Pure Mg and Mg-30Y master alloy). It have been used these based
materials due their applicability in automotive and aerospace domains. Kula et al. highlighted that the presence of Yttrium in magnesium alloys lead to hardness and ductility improvement [16].

There have been performed morphological analyses of deposited layers, phase identification using XRD determination and also identification of some mechanical characteristics using scratch and indentation method.

2. Materials and methods
Ni-Cr-Fe, respectively Cr$_3$C$_2$-NiCr coatings were obtained using atmospheric plasma spray facility (SPRAYWIZARD-9MCE, Sultzer-Metco, USA/ 9MB spraying gun) with typically particle size between 10-90 microns. The substrates of these coatings used in this present research were pure magnesium and a Mg-30Y master alloy (HB New Material, Changsta China). In table 1 are highlighted the technological parameters used for the coating process.

Morphological and microstructure have been determined using a LFD (large field detector), respectively dual BSD (backscattered detector) on a scanning electron microscope SEM Quanta 200 3D DUAL BEAM. For phase analysis, XRD analysis was carried out using an X’Pert Pro MPD diffractometer, with a copper X-ray tube – Kα: 1.54 Å, in 2Theta scanning range of 20º-90º. Apparent coefficient of friction, hardness and elastic modulus were measured using CETR UMT-2 Tribometer. Micro-scratch analysis parameters consist in: a constant load of 5N for a distance of 4 mm, for a single determination, on a flat sample of 20mm x10mm sample. Also, for the indentation determination, there were performed three determinations for each sample with a constant load of 5 N and using a metallic indenter.

Table 1. Technological parameters used for coating process.

| Powder Gun | Ar Pressure (psig) | Ar Gas flow (SCFH) | H$_2$ Pressure (psig) | H$_2$ Gas flow (SCFH) | Electric DC (A) | Electric DC (V) | Electric Carrier gas flow (SCFH) | Air Pressure (psig) | Rate (lb/h) | Spray distance (inch) |
|------------|--------------------|--------------------|----------------------|----------------------|----------------|----------------|-----------------------------|-------------------|--------------|----------------------|
| Cr$_3$C$_2$-NiCr coating | 75 | 111 | 50 | 10 | 500 | 60-70 | 13.5 | 20 | 5.5 | 2.25 |
| Ni-Cr-Fe coating | 75 | 90 | 50 | 20 | 500 | 70-80 | 13.5 | 20 | 4 | 9 |

3. Results and discussions

3.1. Structural analysis

3.1.1. Scanning electron microscopy. Microstructure and morphology aspects of the ceramic coatings are presented in figures 1-4. In figure 1 (a-f) is presented the specific deposition aspects of APS method of a Cr$_3$C$_2$-NiCr, respectively Ni-Cr-Fe based coatings.

These ceramic layers have on the surface an average roughness with the presence of some small unmolded particle, few micropores and some microcracks, due the recovery process of thermal deposition. Different scale images present that in both cases an adherent coating with the substrate.

In figures 2-4 are shown cross-section images (a-f) for Cr$_3$C$_2$-NiCr / Ni-Cr-Fe layers. Ceramic coatings have the appearance of columnar grains called "splats", due the layer by layer spraying technique and having an average thickness of the splat of 3 microns (figure 4).

The average thickness of coatings are presented in figure 2(a, e), respectively in figure 3 (a, d) which is aproximatively 200 µm for Cr$_3$C$_2$-NiCr coating and respectively 150 µm for Ni-Cr-Fe coating.
3.1.2. XRD analysis. XRD diffraction analyses are shown in Figure 5. The diffraction analysis highlighted Cr-based compounds: \( \text{Cr}_3\text{C}_2 \), \( \text{Cr}_7\text{C}_3 \), \( \text{Cr}_3\text{Ni}_2 \), \( \text{Cr}_{0.1}\text{Ni}_{0.9} \), \( \text{Cr}_7\text{Ni}_3 \) having both orthorhombic and tetragonal crystal structure.

For the coatings, \( \text{Cr}_3\text{C}_2 \) and \( \text{Cr}_7\text{C}_3 \) are the predominant phases around at 44.22° 2θ angle. \( \text{Cr}_3\text{Ni}_2 \) and \( \text{FeNi}_3 \) are the minority diffraction peaks of the coatings around 51.67° 2θ angle and 76.24° 2θ angle, with a tetragonal, respectively cubic structure. Base materials consist three types of compounds: \( \alpha\)-Mg (hexagonal structure) and MgY, respectively Mg24Y5 (cubic structure). Lattice parameters of all identified compounds are shown in table 2.
Table 2. Lattice parameters of Cr$_3$C$_2$-NiCr / Ni-Cr-Fe coatings / Base materials (Pure Mg / Mg-30Y).

| Compound                  | Space Group | Crystal system | a (Å) | b (Å) | c (Å) | α (º) | β (º) | γ (º) | Cell volume (10$^6$ pm$^3$) |
|---------------------------|-------------|----------------|-------|-------|-------|-------|-------|-------|-----------------------------|
| Cr$_3$C$_2$-NiCr coating  | Pnam        | Orthorhombic   | 5.5250| 11.4680| 2.8266| 90    | 90    | 90    | 179.10                      |
| Cr$_3$C$_2$               | Pnma        | Orthorhombic   | 4.5260| 7.0100| 12.1420| 90    | 90    | 90    | 385.23                      |
| Cr$_3$Ni$_2$              | P42/mmm     | Tetragonal     | 8.8200| 8.8200| 4.5800| 90    | 90    | 90    | 356.29                      |
| Cr$_7$Ni$_3$              | P42/mmm     | Tetragonal     | 8.7100| 8.7100| 4.4900| 90    | 90    | 90    | 340.63                      |
| Cr$_{0.1}$Ni$_{0.9}$      | P-3m1E      | Cubic          | 3.5350| 3.5350| 3.5350| 90    | 90    | 90    | 44.17                        |
| Ni-Cr-Fe coating          |             |                |       |       |       |       |       |       |                             |
| Cr$_{1.12}$Ni$_{2.88}$    | Fm-3m       | Cubic          | 3.5400| 3.5400| 3.5400| 90    | 90    | 90    | 44.36                        |
| FeNi$_3$                  | Fm-3m       | Cubic          | 3.5556| 3.5556| 3.5556| 90    | 90    | 90    | 44.95                        |
| Cr$_{0.1}$Ni$_{0.9}$      | Fm-3m       | Cubic          | 3.5350| 3.5350| 3.5350| 90    | 90    | 90    | 44.17                        |
| Pure Mg                   | Mg          | P63/mmc        | 3.2088| 3.2088| 5.2099| 90    | 90    | 120   | 46.46                        |
| MgY                       | Pm-3m       | Cubic          | 3.7900| 3.7900| 3.7900| 90    | 90    | 90    | 54.44                        |
| Mg$_{24}$Y$_5$            | I-43m       | Cubic          | 11.2500| 11.2500| 11.2500| 90    | 90    | 90    | 1423.83                     |

Figure 5. XRD patterns of Cr$_3$C$_2$-NiCr / Ni-Cr-Fe coatings.

3.1.3. Scratch and micro-indentation analysis. Figure 6 presents SEM images of scratch test of Cr$_3$C$_2$-NiCr coatings (a, b), respectively Ni-Cr-Fe coatings. The apparent coefficient of friction for the Cr$_3$C$_2$-NiCr layer is in the range of 0.55. Same analyses were carried out for the second layer, respectively Ni-Cr-Fe, which presented an apparent coefficient of friction around 1.25.

![Scratch image](image_url)

Figure 6. SEM images of scratch test for Cr$_3$C$_2$-NiCr coatings (a,b), Ni-Cr-Fe coatings (c,d).

SEM images showed adherent layers without removing to the base material. It have been observe a small number of microcracks in both of the coatings, resulting less fragile layers. Coatings present similar values of the stiffness between 3.589 and 3.953 N/μm. The results of microindetation test have shown a depth of between 6.15 μm and 16.33 μm for Cr$_3$C$_2$-NiCr layer, respectively 8.46 μm and 18.91 μm for the second, without fracturing. It can be concluded that both coatings have high.
resistance to fracturing, a fact confirmed by hardness values and coefficient of friction apparently values presented in table 3 and figure 7. Also the hardness is influenced by the substrate, Ytrrium has superior behavior in magnesium alloys [14].

Figure 7. Some mechanical characteristics obtained by scratch and microindentation analysis: a) Hardness/COF; b) Stiffness; c) Young modulus.

Young modulus is influenced by the substrate characteristics. The coatings performed on pure magnesium substrate show similar values between 22.79 GPa and 23.37 GPa and also the coatings deposited on Mg-30Y revealed higher values of elastic modulus between 33.17 GPa and 38.34 Gpa.

Table 3. Some mechanical properties of Cr3C2-NiCr / Ni-Cr-Fe coatings on Pure Mg, respectively Mg-30Y substrate.

| Coating Type                  | COF  | Hardness (GPa) | Young Modulus (GPa) | Stiffness(N/μm) |
|-------------------------------|------|----------------|---------------------|-----------------|
| Cr3C2-NiCr / Pure Mg          | 0.515| 0.239          | 22.795              | 3.630           |
| Cr3C2-NiCr / Mg-30Y           | 0.629| 0.683          | 38.343              | 3.701           |
| Ni-Cr-Fe / Pure Mg            | 1.298| 0.217          | 23.375              | 3.935           |
| Ni-Cr-Fe / Mg-30Y             | 1.253| 0.513          | 33.172              | 3.589           |

4. Conclusions
The aim of this research was to investigate the morphological appearance, X-ray diffraction, micro-scratch and micro-indentation aspects on Cr3C2-NiCr / Ni-Cr-Fe coatings on Pure Mg, respectively Mg-30Y substrate. These coatings confirm the presence of “splat” particles with an average size of 3 microns. Also, on the surface of the coatings it has been observed typically plasma deposition morphology like, few microcracks, unmolded particles etc. X-Ray analysis showed the presence Cr-based compounds: Cr7C2, Cr7C3, Cr3Ni2, Cr0.1Ni0.9, Cr3Ni3 having both orthorhombic and tetragonal crystal structure as predominant phase. Also was identified as secondary phase Cr2Ni3 and FeNi3, with a crystal structure of tetragonal and cubic type. Young modulus is influenced by the substrate
characteristics. The coatings performed on pure magnesium, respectively Mg-30Y substrate show similar values. Also, the apparent COF of Cr$_3$C$_2$-NiCr is lower than COF of Ni-Cr-Fe coatings. Research will continue with future experiments in order to evaluate the coatings characteristics for different types of applications.

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