Modulation of electrical potential and conductivity in an atomic-layer semiconductor heterojunction

Yu Kobayashi1, Shoji Yoshida2, Ryuji Sakurada2, Kengo Takashima3, Takahiro Yamamoto3,4, Tetsuki Saito5, Satoru Konabe5, Takashi Taniguchi6, Kenji Watanabe6, Yutaka Maniwa1, Osamu Takeuchi2, Hidemi Shigekawa2 & Yasumitsu Miyata1,7

Semiconductor heterojunction interfaces have been an important topic, both in modern solid state physics and in electronics and optoelectronics applications. Recently, the heterojunctions of atomically-thin transition metal dichalcogenides (TMDCs) are expected to realize one-dimensional (1D) electronic systems at their heterointerfaces due to their tunable electronic properties. Herein, we report unique conductivity enhancement and electrical potential modulation of heterojunction interfaces based on TMDC bilayers consisted of MoS2 and WS2. Scanning tunneling microscopy/spectroscopy analyses showed the formation of 1D confining potential (potential barrier) in the valence (conduction) band, as well as bandgap narrowing around the heterointerface. The modulation of electronic properties were also probed as the increase of current in conducting atomic force microscopy. Notably, the observed band bending can be explained by the presence of 1D fixed charges around the heterointerface. The present findings indicate that the atomic layer heterojunctions provide a novel approach to realizing tunable 1D electrical potential for embedded quantum wires and ultrashort barriers of electrical transport.

Semiconductor heterojunctions are vital components in modern solid-state physics and have applications in electronics and optoelectronics. As an example, conventional three-dimensional heterojunctions of crystalline semiconductors, such as GaAs/AlGaAs, can generate a high-mobility two-dimensional electron gas (2DEG) at their two-dimensional interface1–3. The realization of clean 2DEG in this manner has led to the discovery of exotic physical phenomena, including fractional quantum Hall effects4,5, and to practical applications in the channels within high-performance field-effect transistors4. A high mobility 2DEG and interface superconductivity have also been found at the heterointerfaces of oxides6,7. Recently, similar heterojunctions consisting of so-called atomic layers have been fabricated in two-dimensional crystals, and have attracted much attention based on their unique low-dimensional structures and physical properties8–18. The in-plane, lateral heterojunctions employed in initial studies were composed of metallic graphene and insulating hexagonal boron nitride (BN) monolayers8–12. Atomic layer-based semiconductor heterojunctions such as these are expected to represent ideal systems for the realization of novel confined electronic systems at one-dimensional (1D) interfaces, and should lead to novel physics and electronic/optoelectronics discoveries.

Several groups have reported the synthesis of semiconductor heterojunctions based on transition metal dichalcogenides (TMDCs) by vapor phase growth techniques13–17. TMDCs are ideal materials for semiconductor heterojunctions because of their unique compositions, polytypes and layer number-dependent electronic properties19,20. In early studies, these heterojunctions were composed primarily of Mo- and W-based semiconductor TMDCs, such as MoS2, MoSe2, WS2 and WSe2 monolayers13–15. In a previous paper, we demonstrated the use of similar lateral heterostructures employing monolayers of Mo1−xWxS2 alloys, which have composition-dependent tunable bandgaps16,17. These earlier studies generated band diagrams by scanning tunneling microscopy/
spectroscopy (STM/STS), and showed that such monolayer heterointerfaces result in staggered type-II heterojunctions in which there is no potential barrier for carrier confinement. Despite these significant earlier works, there have been no reports to date of the conducting 1D interface and 1D confining potential/potential barriers associated with these TMDC heterojunctions, likely because these prior works have focused solely on monolayer-based heterojunctions.

In the present study, we observed conductivity enhancement and unique band bending at the interface of TMDC bilayer-based heterojunctions. These heterojunctions were composed of WS₂ bilayers and vertically stacked WS₂/MoS₂ heterostructures, grown by two-step chemical vapor deposition (CVD). The conductivity enhancement was assessed by conducting atomic force microscopy (c-AFM). Furthermore, STM/STS measurements demonstrated an upshift in both the valence and conduction band edges, as well as band-gap narrowing around the heterointerface. These results indicate that the highly tunable electronic properties of TMDCs provide an ideal system allowing the realization of tunable 1D electronic systems at the heterointerfaces between atomic layers.

Results
High-temperature CVD was used to prepare high-quality TMDC samples on graphite and boron nitride flakes, as reported in our previous paper. This process forms mono- and multilayer grains of MoS₂ and WS₂ with sizes around 10 μm. As shown in Fig. 1a, WS₂ (or MoS₂) grains are initially synthesized on the graphite, after which the second growth process of another TMDCs is conducted with firstly-grown grains after changing the reaction chamber to avoid alloy formation, in agreement with a previous report by Li et al. In addition to the formation

Figure 1. Growth and structures of three types of heterojunctions based on WS₂ and MoS₂. (a) Optical images and illustrations for a sample before and after MoS₂ growth around WS₂ grains on graphite. (Note: in the structural model, cyan, red and yellow correspond to W, Mo and S atoms, respectively.) (b) Illustration of a growth model for a bilayer heterojunction based on bilayer WS₂-stacked WS₂/MoS₂. (c) Structures and measurement setup for the three heterojunctions studied in the present work.
of a conventional monolayer heterojunction, we also found that a second TMDC layer can simultaneously grow both from the grain edges and on the grain surfaces, leading to the formation of bilayer TMDC-based heterojunctions, as shown in Fig. 1b. We note that only the initial monolayers formed on the graphite or BN tend to survive, and that these same monolayers grown on SiO2 or sapphire substrates easily degrade during the second growth process at 1100 °C. These findings indicate that the stability of WS2 and MoS2 is sensitive to the growth substrate, and is a critical factor in the second growth. In the present work, we investigated three types of heterojunctions, including monolayer WS2 and MoS2 (1L WS2-MoS2), bilayer WS2 - stacked WS2/MoS2 (2L WS2-WS2/MoS2) and monolayer and bilayer WS2 (1L-2L WS2), using c-AFM and STM/STS, as shown in Fig. 1c.

A model of the heterostructures composed of a triangular WS2 monolayer with lateral MoS2 is shown in Fig. 1a. In this sample, the inner and outer parts correspond to WS2 and MoS2 monolayers, respectively, as confirmed by AFM images and a height profile (Fig. 2a), as well as the photoluminescence (PL) intensity maps (Fig. 2b–d) and Raman data (Supplementary Fig. S1). The PL maps show that the WS2 is primarily located in the inner part of the crystal, whereas the MoS2 is formed around the inner WS2. Almost the same structure is indicated by the Raman intensity maps of WS2 and MoS2. In the PL and Raman spectra (Supplementary Fig. S2), both the excitonic PL peaks and vibrational modes of WS2 and MoS2 are observed around the interface, with no peak shift or broadening caused by alloy formation. Furthermore, the PL peaks exhibit symmetric Lorentzian profiles and narrow linewidths, with full width at half maximum (FWHM) values of 20 and 47 meV for the WS2 and MoS2 (Supplementary Fig. S3), respectively, indicating high-quality samples.

In addition to the optical measurements, the electrical properties of the monolayer WS2/MoS2 heterojunction were investigated, using c-AFM in air. To obtain a current map, the conducting tip was scanned along a TMDC specimen grown on a graphite substrate, which served as an electrode (Fig. 1c). As shown in Fig. 2e,f, this technique allows visualization of the difference in the vertical electrical conductivities between the inner WS2 and the outer MoS2. The enlarged current image indicates a straight, steep heterojunction interface (Fig. 2f). Interestingly, the current profile reveals that the current value abruptly changes from 135 nA in the MoS2 to 115 nA in the WS2 within a region of 8 nm around the interface (Fig. 2g). In contrast, there are no differences in the topographic image and height profile in the same area (Supplementary Fig. S4). The continuous change in current suggests that the bandgap gradually changes in the vicinity of the heterointerface, and that there is no large band offset as reported in our previous STM/STS study of monolayer-based lateral heterostructures.

A similar current profile was obtained for the 1L-2L WS2 heterojunction. As shown in Supplementary Fig. S5a,b, additional triangular WS2 monolayers were occasionally grown on the surfaces of large triangular WS2 monolayers during the CVD process. The current image and profile of the grain also demonstrate a difference in conductivity between the 1L and 2L WS2 (Supplementary Fig. S5c,d). In the profile, the current changes continuously from 140 nA in the 2L region to 80 nA in the 1L region. This strongly suggests that the spatial changes in the bandgap of the 1L-2L WS2 heterojunction are similar to those that occur in monolayer heterojunctions and that the edge state of the top layer has only a limited effect on the present current measurements in air.

In contrast, drastic current enhancement at the heterointerface is observed in the case of the bilayer-based heterojunctions. In the present bilayer samples, WS2 was grown from the edge and on the surfaces of monolayer
MoS<sub>2</sub> grains, as illustrated in Figs 1b and 3a. Both the AFM image and the height profile show triangular crystals approximately 5 μm in size and with heights of 1.4 nm, corresponding to the thickness of the bilayer TMDC (Fig. 3b). The Raman intensity maps (Fig. 3c–e) and spectra (Supplementary Fig. S6) of the same crystal exhibit a MoS<sub>2</sub> peak only in the inner part of the crystal. Conversely, WS<sub>2</sub> peaks were generated for both the inner and outer parts and have a particularly high intensity in the outer part. The PL map and spectra (Supplementary Fig. S7) indicate only monolayers around the corners, which generate sharp PL peaks, while no signal is observed for the bilayer part grown on graphite. This can be understood as a result of the indirect bandgap nature of both the 2L WS<sub>2</sub> (refs 22,23) and MoS<sub>2</sub>/WS<sub>2</sub> vertical heterostructures (refs 24–26), and the presence of fast non-radiative decay processes on the graphene surface<sup>21</sup>. These results demonstrate the formation of lateral and vertical heterostructures of MoS<sub>2</sub> and WS<sub>2</sub>, as presented in Fig. 3a. Importantly, these heterostructures can be regarded as the lateral heterojunction of the outer 2L WS<sub>2</sub> and inner WS<sub>2</sub>/MoS<sub>2</sub> vertically stacked heterostructures.

Unlike the current images of the 1L MoS<sub>2</sub>/WS<sub>2</sub> and 1L-2L WS<sub>2</sub>, the bilayer grain exhibits black lines with a triangular shape along the interface (Fig. 3f,g), indicating that a significant current can flow across the bilayer from the tip to the graphite substrate. For the in-plane parts, the inner triangle regions exhibit somewhat higher current values than the outer parts of the sample, which is consistent with the different electronic structures of these portions, as observed in the PL spectra. Interestingly, the current is enhanced by several tens of times over a 10 nm
Discussion

Finally, we discuss the origin of the observed band profile around the heterointerface of bilayer samples. Possible origins could include an interface state due to Mo-S-W bonds, different electron affinities, lattice strain, impurity doping, structural defects, changes in interlayer interaction, and piezoelectric polarization. From density functional calculations of the local density of state, we found that no mid-gap states are formed in the bandgaps around the heterointerface as compared to the in-plane bilayers. We note that the same results have been obtained with other bilayer grains (Supplementary Fig. S9). These conducting 1D interfaces are a unique feature in the present bilayer heterojunctions, and are observed for the first time ever in the present study.

To obtain additional insights into the origin of the conducting interface, STM/STS studies of the bilayer heterostructures were performed. Figure 4a shows a high-resolution STM image around the interface of the bilayer heterojunction. In this image, the left and right sides correspond to 2L WS2 and WS2/MoS2, respectively. This image shows a clear lattice composed of a surface WS2 layer and no structural defects around the interface. In Fig. 4b, FFT images of both the left and right sides of the STM image demonstrate six-fold symmetry spots with the same orientation. The wide-area STM image shows a dip of approximately 0.1 nm along the interface.

As clearly shown in these maps and spectra, the present heterojunctions have a bandgap of approximately 1.5 eV (Fig. 4e), which is less than those of the 2L WS2 (2.1 eV) and WS2/MoS2 (2.0 eV). The mismatch between the lowest PL peak and the bandgap estimated by STM/STS is probably caused by several factors including the large exciton binding energies and the different measurement conditions such as temperature and substrates. Spatial changes in the band profile are observed within a 20 nm width zone around the interface as thresholds in the map using a dI/dV color scale (Fig. 4f). The E_CBM undergoes a larger upshift than the E_VBM, thus narrowing the bandgap. As clearly shown in these maps and spectra, the present bilayer sample forms a type II semiconductor heterojunction with staggered bandgaps. The small bandgap of the WS2/MoS2 is also consistent with the present PL results (Supplementary Fig. S8) and with theoretical predictions based on first principles calculations. These results indicate that the bandgap narrowing and/or the upshift in E_VBM lead to current increases around the interface during c-AFM measurements. In particular, the confining potential in the valence band could result in the accumulation of charge carriers along the interface and generate a 1D hole gas. In contrast, the potential barrier in the conduction band would allow control of the electron transport properties across the heterointerface by electric gating and thus would have applications to ultra-short channel switching devices.

Figure 4. STM and STS study of a heterojunction based on bilayer TMDCs. (a) High-resolution STM image around the bilayer-based heterointerface (V_T=+1.4 V, I=10 pA). (b) FFT images of left and right sides of (a). (c) Wide-area STM image around the bilayer heterointerface (V_T=+1.2 V, I=10 pA). (d) dI/dV maps at different sample bias voltages of V_T=0.7, −1.0, −1.5 and −1.8 V in the same area as (e) (V_T=+1.2 V, I=10 pA). (e) Averaged dI/dV spectra obtained from the regions indicated by blue (2L WS2), red (WS2/MoS2), and green (interface) squares in (c). (f) dI/dV color scale map calculated from the spatially resolved STS spectra along the white dotted line in (c).
of the 2L WS₂ and MoS₂/WS₂ heterointerface (Fig. 5a). We also note that no large band offset has been observed for either the E_VM or E_CB in the STS analyses, which would be expected to be induced by a difference in electron affinity. We therefore propose that the band bending results from a combination of piezoelectric-charge-induced band upshift and strain-induced bandgap narrowing as a possible candidate, as shown in Fig. 5b, for the reasons detailed below.

In the case of the present bilayer interface, the E_CB undergoes a remarkable upshift around the interface. This suggests that interlayer coupling could be one of the major factors affecting the bandgap narrowing because, for bilayer TMDCs, the E_VM at the Γ point is predicted to be sensitive to the interlayer interactions. The interlayer coupling can likely be changed by lattice strain, which generally has a maximum value at the heterointerface when the two materials have different lattice constants. In the present bilayer samples, the lattice constants and thermal expansion coefficients are still unknown but could have different values compared with those of monolayer samples. Importantly, the presence of interface strain can also explain the upshift in E_CB in response to piezoelectric charges. Recently, the piezoelectric effect in TMDCs has been revealed by theoretical and experimental works. Calculations predict that the 1L MoS₂ and WS₂ will have piezoelectric coefficients of 364 and 247 pC/m, respectively. Using these values, the interface polarization, \( \mathcal{P}_i \), can be readily estimated to be 117\( \varepsilon_{11} \) pC/m, where \( \varepsilon_{11} \) is the uniaxial strain. In the present work, the value of the interface space charge was evaluated from the measured band profile. Here, we assume that the E_CB of the 2L WS₂ region is up-shifted only by a fixed, negative cylindrical space charge with linear charge density \( \lambda \), as pictured in Fig. 5c, with a spatially uniform dielectric environment. From familiar Gauss's law, the potential distribution can be expressed as

\[
V(r) = \frac{-\lambda}{4\pi\varepsilon_0 a^2} r^2 + V_0, \quad (r \leq a) 
\]

\[
V(r) = \frac{-\lambda}{4\pi\varepsilon_0} \left(1 + 2 \ln \frac{r}{a}\right) + V_0, \quad (r \geq a)
\]

where \( r \) is the distance from the center, \( a \) is the radius of the cylindrical space charge, \( \varepsilon_0 \) is an effective dielectric constant and \( V_0 \) is the potential of the center. We found that this simple relation works well to reproduce the E_CB profile (Fig. 5c) using the fitting parameters 1.0 nm for \( a \), 4.4 pC/m for \( \lambda/\varepsilon \) and 0.93 V for \( V_0 \). Note that this logarithmic dependence is a signature of 1D linear charges, which are different from conventional 2D plane charges. It is noteworthy that the resulting linear charge density is comparable to the value expected from piezoelectric polarization, for example, \( P_i \) becomes 6 pC/m when \( \varepsilon_{11} = 0.05 \), if the interface has several percent lattice strain and if we use the calculated \( \varepsilon \) of 5.2 for bilayer MoS₂. Even though the accuracy of used values should be improved in the future, this supports the validity of our proposed model as a possible mechanism for the band bending.

It is also conceivable that the band bending is caused by other extrinsic factors, including the presence of impurities and structural defects such as sulfur vacancies, edges and grain boundaries. Recent STM/STS studies have reported the presence of mid-gap states at the heterointerface of 1L and 2L WS₂ (ref. 30), and the bandgap narrowing and the upshift of E_CB for the grain boundaries of 1L MoS₂ (ref. 31). A striking difference of the present work is that band bending was observed over a rather wide range of approximately 20 nm. In contrast, in previous studies, the local changes in E_VM and E_CB occur within an area of around 5 nm. This difference suggests that the present band bending is derived from another mechanism and is most likely affected by long-range lattice strain and/or fixed charges. Because the effects of such extrinsic factors are still unclear, further studies including the analysis of interface structures are necessary to reveal the origin of observed band bending.

In conclusion, we have observed the conductivity enhancement and band bending of a heterojunction interface composed of two-dimensional semiconductors. In particular, the bilayer heterojunction formed by 2L WS₂ and WS₂/MoS₂ shows a unique band profile, as revealed by STM/STS. The present findings demonstrate that the
atomic-layer heterojunctions provide an ideal system to realize tunable 1D electrical potential at heterointerfaces and should have applications in advanced electronics and opto-electronics. We note that 1D electron/hole gases and liquids exhibiting unique physical phenomena can be generated through carrier doping to the interface. Furthermore, it is suggested that these heterointerfaces allow us to control local confining potential and potential barrier via the piezoelectric effect, and so could be used as future switching devices, piezoelectric devices, and strain sensors. The interface fixed charges can also generate electrical field at the interface, which may be used to manipulate valley degree of freedom in TMDC-based optoelectronic devices. The realization and application of such interface-related phenomena are exciting challenges in the field of atomic layer science.

Methods
Sample preparation. WS₂ and MoS₂ heterostructures were grown on Kish graphite (Type C, Covalent Materials Co.) or BN flakes using CVD, as reported previously. The graphite crystals were mechanically exfoliated onto quartz substrates using Nito tape (SPV–224). During the initial growth process, the substrate was placed in a quartz tube (3 cm diameter, 100 cm in length) together with WO₃ (Aldrich, 99% purity, 15–50 mg) or MoO₂ (Strem Chemicals, 99% purity, 5–15 mg) powder and sulfur flakes (Aldrich, 99.99% purity, 2 g). The quartz tube was then filled with Ar gas at a flow rate of 200 cm³/min. The temperature of the substrate and the WO₃ or MoO₂ powder was gradually increased to the sulfurization temperature (1100 °C) over 60 min using an electric furnace. When the substrate temperature was at the set point, the sulfur was heated at 160–180 °C for 15–30 min to supply sulfur vapor to the substrate, using a second electric furnace. Following the growth process, the quartz tube was immediately cooled using an electric fan. For the second growth, the sample was placed in a quartz boat, with the precursor (WO₃ or MoO₂) and sulfur flakes in a second quartz tube to avoid contamination. When the furnace temperature was at the set point, the sample was immediately introduced into the hot zone from the exterior of the furnace using a magnet in the quartz boat, and so exposed to the sulfur vapor. After the reaction had proceeded for 15–30 min, the sample was immediately cooled.

Characterizations. Optical images were recorded with an optical microscope (Nikon, Eclipse-EV100D). Raman spectra were acquired using a micro-Raman spectroscope (Renishaw, inVia) with a 532 nm excitation laser. Topography and current images of the samples were obtained using atomic force microscopy (AFM, Shimadzu, SPM-9600). The current images were acquired at a bias voltage of −0.1 V. We note that the current is sensitive to the tip conditions because the conductivities between tip and samples are affected by various factors including tip shape and surface impurities of the tip. STM/STS assessments were carried out using a low-temperature STM system (UNISOKU, USM1200) operating at 83 K.

Theoretical calculations. First principles calculations for projected local density of state were performed using the Atomistix Toolkit DFT code (Ver.2015.1) based on the nonequilibrium Green's function method combined with density functional theory. PBE-GGA was adopted as the exchange correlation functional and the basis set was double-zeta-polarized. A norm conserving pseudopotential was used for all atoms. The real-space grid cutoff was 75 Hartree. The geometry and the cell size were optimized until all forces were below 0.001 eV/Å using the Atomistix Toolkit DFT code (Ver.2015.1) based on the nonequilibrium Green's function method.

References
1. Störmér, H. L., Dingle, R., Gossard, A. C., Wiegmann, W. & Sturge, M. D. Two-dimensional electron gas at a semiconductor-semiconductor interface. Solid State Commun. 29, 705–709 (1979).
2. Takashi, M., Satoshi, H., Toshiro, F. & Kazuo, N. A new field-effect transistor with selectively doped GaAs/n-AlGaAs heterojunctions. Jpn. J. Appl. Phys. 19, L225 (1980).
3. Dingle, R., Störmér, H. L., Gossard, A. C. & Wiegmann, W. Electron mobilities in modulation-doped semiconductor heterojunction superlattices. Appl. Phys. Lett. 33, 665–667 (1978).
4. Tsu, D. C., Störmér, H. L. & Gossard, A. C. Two-dimensional magnetotransport in the extreme quantum limit. Phys. Rev. Lett. 48, 1559–1562 (1982).
5. Störmér, H. L., Tsui, D. C. & Gossard, A. C. The fractional quantum Hall effect. Rev. Mod. Phys. 71, S298–S305 (1999).
6. Ohtomo, A. & Hwang, H. Y. A high-mobility electron gas at the LaAlO₃/SrTiO₃ heterointerface. Nature 427, 423–426 (2004).
7. Reyren, N. et al. Superconducting interfaces between insulating oxides. Science 317, 1196–1199 (2007).
8. Levendov, M. P. et al. Graphene and boron nitride lateral heterostructures for atomically thin circuitry. Nature 488, 627–632 (2012).
9. Sutter, P., Cortes, R., Lahiri, J. & Sutter, E. Interface formation in monolayer graphene–boron nitride heterostructures. Nano Lett. 12, 4869–4874 (2012).
10. Miyata, Y. et al. Fabrication and characterization of graphene/hexagonal boron nitride hybrid sheets. Appl. Phys. Express 5, 085102 (2012).
11. Liu, Z. et al. In-plane heterostructures of graphene and hexagonal boron nitride with controlled domain sizes. Nature Nanotech. 8, 119–124 (2013).
12. Liu, L. et al. Epitaxial growth of two-dimensional hexagonal boron nitride templated by graphene edges. Science 343, 163–167 (2014).
13. Gong, Y. et al. Vertical and in-plane heterostructures from WS₂/MoS₂ monolayers. Nature Mater. 13, 1135–1142 (2014).
14. Duan, X. et al. Lateral epitaxial growth of two-dimensional layered semiconductor heterojunctions. Nature Nanotech. 9, 1024–1030 (2014).
15. Huang, C. et al. Lateral heterojunctions within monolayer MoSe₂/WSe₂ semiconductors. Nature Mater. 13, 1096–1101 (2014).
16. Kobayashi, Y., Moro, S., Maniwa, Y. & Miyata, Y. Bandgap-tunable lateral and vertical heterostructures based on monolayer Mo₁₋ₓ WₓS₃ alloys. Nano Res. 8, 3261–3271 (2015).
17. Yoshida, S. et al. Microscopic basis for the band engineering of Mo$_{1-x}$W$_x$S$_2$-based heterojunction. Sci. Rep. 5, 14808 (2015).
18. Li, M.-Y. et al. Epitaxial growth of a monolayer WSe$_2$-MoS$_2$ lateral p-n junction with an atomically sharp interface. Science 349, 524–528 (2015).
19. Chhowalla, M. et al. The chemistry of two-dimensional layered transition metal dichalcogenide nanosheets. Nature Chem. 5, 263–273 (2013).
20. Wang, Q. H., Kalantar-Zadeh, K., Kis, A., Coleman, J. N. & Strano, M. S. Electronics and optoelectronics of two-dimensional transition metal dichalcogenides. Nature Nanotech. 7, 699–712 (2012).
21. Kobayashi, Y. et al. Growth and optical properties of high-quality monolayer WS$_2$ on graphite. ACS Nano, 4056–4063 (2015).
22. Zhao, W. et al. Origin of indirect optical transitions in few-layer MoS$_2$, WS$_2$, and WSe$_2$. Nano Lett. 13, 5627–5634 (2013).
23. Zhao, W. et al. Evolution of electronic structure in atomically thin sheets of WS$_2$ and WSe$_2$. ACS Nano 7, 791–797 (2013).
24. Kośmider, K. & Fernández-Rossier, J. Electronic properties of the MoS$_2$-WS$_2$ heterojunction. Phys. Rev. B 87, 075451 (2013).
25. Terrones, H., López-Urías, F. & Terrones, M. Novel hetero-layered materials with tunable direct band gaps by sandwiching different metal disulfides and diselenides. Sci. Rep. 3, 1549 (2013).
26. Komsa, H.-P. & Krasheninnikov, A. V. Electronic structures and optical properties of realistic transition metal dichalcogenide heterostructures from first principles. Phys. Rev. B 88, 085318 (2013).
27. Duerloo, K.-A. N., Ong, M. T. & Reed, E. J. Intrinsic piezoelectricity in two-dimensional materials. J. Phys. Chem. Lett. 3, 2871–2876 (2012).
28. Wu, W. et al. Piezoelectricity of single-atomic-layer MoS$_2$ for energy conversion and piezotronics. Nature 514, 470–474 (2014).
29. Cheiwchanchamnangij, T. & Lambrecht, W. R. L. Quasiparticle band structure calculation of monolayer, bilayer, and bulk MoS$_2$. Phys. Rev. B 85, 205302 (2012).
30. Zhang, C. et al. Visualizing band offsets and edge states in bilayer-monolayer transition metal dichalcogenides lateral heterojunction. Nature Commun. 7, 10349 (2016).
31. Huang, Y. L. et al. Bandgap tunability at single-layer molybdenum disulphide grain boundaries. Nature Commun. 6, 6298 (2015).
32. Zhang, Y. J., Oka, T., Suzuki, R., Ye, J. T. & Iwasa, Y. Electrically switchable chiral light-emitting transistor. Science 344, 725–728 (2014).
33. Stokbro, K., Blom, A. & Smidstrup, S. Atomistic simulation of a III–V p-i-n junction: Comparison of density functional and tight-binding approaches. International Conference on Simulation of Semiconductor Processes and Devices (SISPAD). 380–383 (2013).

Acknowledgements
This work was supported by JSPS KAKENHI Grant Numbers 15H05412, 26107530, 15H05734 and 15H03523.

Author Contributions
Y.K. prepared all experimental samples and performed optical and AFM measurements. T.S. assisted the sample preparation and optical measurements. S.Y., R.S., Y.K. and Y.Miyata carried out the STM/STS experiments. H.M. and O.T. provided technical assistance. K.T. and T.Y. performed the theoretical calculations. T.T. and K.W. supplied the BN. Y.Miyata developed the concept and supervised the project. Y.K. and Y.Miyata prepared the figures and wrote the paper. S.Y., H.S., T.Y., K.T., S.K. and Y.Maniwa reviewed and corrected the paper. All authors discussed the results and commented on the manuscript.

Additional Information
Supplementary information accompanies this paper at http://www.nature.com/srep

Competing financial interests: The authors declare no competing financial interests.

How to cite this article: Kobayashi, Y. et al. Modulation of electrical potential and conductivity in an atomic-layer semiconductor heterojunction. Sci. Rep. 6, 31223; doi: 10.1038/srep31223 (2016).

This work is licensed under a Creative Commons Attribution 4.0 International License. The images or other third party material in this article are included in the article’s Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder to reproduce the material. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/

© The Author(s) 2016