Prediction of Europium Retention in Perovskite: Potential Candidates for an Engineering Barrier in the Disposal of Radioactive Waste

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Perovskites, such as tausonite, are crystalline metal oxides with excellent optical and photocatalytic properties and have also been used successfully in the retention of metals, simulating the isotopes of uranium and plutonium. In this work, different pseudo-order and thermodynamic models were studied to achieve the prediction of the sorption of Eu³⁺ (chemical analogous for actinides) in tausonite. The effects of gamma irradiation and temperature on the structural characteristics of the material were determined, as an additional step in the evaluation of material as an engineering barrier in the disposal of radioactive waste. The results obtained show that the tausonite is resistant to the gamma irradiation and thermal energy. Likewise, it was possible to determine that europium sorption occurs through an exothermic and spontaneous reaction, as well as through the formation of surface complexes, where Eu³⁺ ions bind to sites on the tausonite by dipole-dipole interaction. Furthermore, it was shown that the sorption mechanism is influenced by diffusive phenomena, which participate in the formation of surface complexes. Additionally, a new sorption model with respect to pH was proposed, which allowed determining the physical parameter $\pi$. The evidence obtained suggests that $\pi$ is a physical parameter that relates pH to an optimal value and could explain the equilibrium between the surface complexes that tausonite forms with europium. Likewise, the evidence suggests that 50 kg of tausonite would have the capacity to retain at least 26.59 g of alpha-emitting radionuclides, equivalent to a waste package (900 kg) with a maximum activity of 4000 Bq/g.

1. Introduction

Perovskites are crystalline metal oxides with the general formula ABO₃, in which A can be rare earth (mainly lanthanides) or an alkali or alkaline earth metal and B is a transition metal [1, 2]. These oxides are excellent light absorbers and their properties include tunable band gaps, low exciton binding energy, high carrier mobility, long diffusion length, and so on [3], and, also, they have a narrow band gap (semiconductor), for which they could be considered as good candidates for thermolectric generators [4, 5]. Among the different structures of perovskites, those of the ATIO₃ type (A = Ca, Ba, Sr, etc.) are the most interesting due to their optical and electronic properties [6, 7]. Perovskites have high tolerance to metal substitution in their crystalline structure, excellent thermal stability, and resistance to sintering of substituted metals [8]. These characteristics have allowed them to be used to create different types of materials by
partial substituting of metals, obtaining modified perovskites that have been used in various photocatalytic applications [9–11]. On the other hand, perovskites, such as taursonite, have been successfully used in the retention of $\text{U}^{6+}$ and $\text{Eu}^{3+}$, simulating the different isotopes of uranium and actinides ($\text{Ac}^{3+}$), which are produced in the burning and reprocessing of nuclear fuels [12, 13]. These studies have shown that perovskites are insoluble in water and have high chemical stability under acidic and alkaline pH conditions and that retention of $\text{U}$ and $\text{Eu}$ is carried out by the formation of inner-sphere surface complexes at pH values low and high, respectively [14, 15]. These results reveal the potential of perovskites as materials that could be used as engineering barriers in the disposal of radioactive waste.

The IAEA defines the term “disposal” as the emplacement of radioactive waste into a facility or location with no intention of recovering it, when the objective of the radioactive waste disposal is to protect the environment and people from the potential radiological consequences due to the release and dispersal of radionuclides [16]. Therefore, disposal involves (a) isolating waste from the biosphere and people, reducing the risk of human intrusion, and (b) containing the wastes, inhibiting, reducing, and delaying the migration of radionuclides into the biosphere [16, 17]. The physical and chemical properties of engineering barriers provide containment of radionuclides. This containment consists of impermeability to water; limited corrosion, dissolution, leaching rate, and solubility; retention of radionuclides; and delayed migration of radionuclides [16].

Typically, the evaluation of materials used as engineering barriers consists of determining their physical and chemical properties, including retention and transport of radionuclides, as well as determining the resistance to degradation of these materials with sources of thermal energy or gamma radiation [16, 17].

There are different mathematical models to assess the retention of radionuclides; these are usually classified into two large groups: (a) microscopic or thermodynamic and (b) macroscopic or empirical. The first group explains the physical chemical and thermodynamic phenomena of the medium (pH, temperature, pressure, redox potential, electrostatic potential, etc.) that influence the retention of sorbate. On the other hand, empirical models simplify these phenomena and focus on establishing the retention kinetics and the affinity of the solid for sorbate.

In the study of the retention capacities of materials used in engineering barriers, it is recommended to use a combination of macroscopic and microscopic models, since they allow estimating important constants and parameters, such as kinetics and sorption capacity, concentration of surface sites, dependence of the surface sites with respect to pH, the formation constants of the surface site, the identification of the surface complexes formed, and the respective complex formation constants, among others. These elements are important in the study of the behavior of engineering barriers for the containment of radionuclides and are part of the “safety case” approach, during the institutional control stage of radioactive waste disposal [18].

Considering that, in previous studies the concentration of surface sites, the dependence of surface sites with respect to pH, the surface site formation constants, identification of surface complexes, and the complex formation constants were established in taursonite [14, 15]. In this work, the study of different macroscopic pseudo-order models ($0.5 \leq n \leq 2$) based on the reduced Van’t Hoff equation is proposed for the prediction of the sorption of $\text{Eu}^{3+}$ (actinides, $\text{Ac}^{3+}$, and simulating) on taursonite. At the same time, the effects of gamma irradiation (500, 1000, 5000, and 12000 kGy) and temperature (25–1273.15 K) on the structural characteristics of the material were determined, as an additional step in the evaluation of taursonite as an engineering barrier in the disposal of radioactive waste. Likewise, the use of optimal modelling of europium retention in the material seeks a new phenomenological point of view that contributes to the discussion and specific interpretation of the new parameters. Therefore, this work also seeks to contribute to the environmental and global challenge that radioactive waste management represents, evaluating materials that can be used successfully in the isolation and containment of radioactive waste.

2. Materials and Methods

2.1. Materials. All reagents used in the experimental tests were analytical grade. Strontium titanate (taursonite) of 99.5% purity (Sigma-Aldrich) was used as an engineering barrier (sorbent). Likewise, potassium nitrate (Sigma-Aldrich, ≥99%) was used as the background salt. Europium nitrate (Aldrich, 99.9%) was used for the experimental sorption tests on perovskite. All solutions were prepared with deionized water under nitrogen flow.

2.2. Gamma Irradiation and Effect of Temperature on Taursonite. Gamma irradiation of perovskite was performed to determine the structural changes due to gamma radiation. Exposure to gamma radiation was performed in a JS–6500 industrial irradiator containing a $^{60}\text{Co}$ source of $28971.56\text{TBq}$ (1 July 2018). The solid was placed in aluminum containers (in duplicate) and subjected to a dose of 500 kGy, 1000 kGy, 5000 kGy, and 12000 kGy, applying a dose rate calculated between 14.09 kGy/h and 14.37 kGy/h (8 September 2018).

The effects of gamma radiation on the structure of taursonite were evaluated by X-ray diffraction (XRD) and scanning electron microscopy (SEM). Taursonite XRD patterns were obtained on a Siemens D-5000 diffractometer using Cu Kα1 radiation. At the same time, a Philips XL-30 scanning electron microscope coupled with energy dispersive X-ray spectroscopy (EDS) was used to study the morphology of the samples. In addition, the effects of temperature on taursonite were evaluated by thermogravimetric analysis (TGA).

Thermogravimetric analyses were performed with a heating rate of 283.15 K/min up to 1273.15 K/min, flowing $\text{N}_2$ (Q600 Calorimeter, T.A. Instruments-Waters LLC) to determine the loss of mass at high temperatures. In addition,
the tausonite samples were calcined at 1023.15 K (750°C) and
1173.15 K (900°C) and analyzed by IR spectroscopy. The IR
spectrum of the solid was recorded in the range of
400–400 cm−1 with a Perkin Elmer 1600 FT-IR spec-
trometer. Samples of the material were mixed with KBr
(infrared grade) in a ratio of 1:200.
Finally, the specific area, pore volume, and mean pore
diameter of the material were determined by interpreting the
adsorption-desorption isotherms for N2 by the standard
multipoint BET method (Brunauer–Emmett–Teller) in a
Belsorp Max equipment (Japan INC).

2.3. Sorption Experiments. Europium sorption experiments
on tausonite were carried out using a solution of 1 × 10−4 M
europium nitrate in 0.5 M potassium nitrate at room tem-
perature under a nitrogen atmosphere. Sorption kinetics and
modelling of europium retention as a function of pH were
performed in batch. Finally, the quantification of europium
was performed by luminescence in a Jobin Yvon Horiba
Fluorolog with a xenon lamp (450 W). 500 µL samples of the
supernatant were excited at 397 nm and europium emission
lines were attributed to two possible reasons. First, due to a
mutual torsion of tetrahedra under the effect of gamma
radiation and the physical darkening of all the samples
because of the gamma dose received. On the other hand,
the diffraction patterns of the irradiated samples were compared
with the standard number card 35–0734. Figure 1 shows the
main diffraction peaks of the samples at 2θ = 23.0°, 32.4°,
40.4°, 57.8°, 67.8°, 77.2°, and 79.9° belonged to 100, 110, 111,
200, 211, 220, and 310, which correspond to tausonite. The
diffraction patterns show that in all cases, the material
presents an intensity of high diffraction peaks, which in-
dicates a high crystallinity. Furthermore, no phase changes
shifts or significant changes in diffraction patterns are ob-
served. However, a decrease in the intensity of the diffraction
peaks was observed. The changes observed in the intensity
lines can be attributed to two possible reasons. First, due to a
mutual torsion of tetrahedra under the effect of gamma
radiation and the physical darkening of all the samples.

2.4. Sorption Kinetics Modelling. The europium sorption
isotherms as a function of time were developed by estimating
the concentration of europium nitrate retained in tausonite
at different contact times (1, 2, 3, 4, 5, 7, 16, 18, 20, 22, 24,
and 48 h). Sorption kinetics were modelled using pseudo-order
expressions (0.5 ≤ n ≤ 2), based on the reduced Van’t Hoff
equation (see equation (1)) and compared with empirical
pseudo-first-order model (Lagergren and Elovich equation),
pseudo-second-order, and intraparticle diffusion. Van’t Hoff
relationship was given by

\[
dC_i(t) \over dt = k_mC_i(t),
\]

where \( C_i(t) \) is the sorbate concentration as a function of time,
\( n \) represents the order of the reaction, and \( k \) is the sorption
kinetic constant. The sorbate concentration retained in the
material is usually expressed as the sorption capacity \( q_i \), as
presented in

\[
q_i = \frac{(C_i - C_i(0))}{m}V,
\]

where \( m \) is the mass of the sorbent, \( V \) is the volume of the
sorbate solution, and \( C_i \) is the initial concentration of the
sorbate. The general equations and their linear representa-
tion used were expressed in terms of \( q_i \) and \( q_e \) (sorption
capacity at equilibrium) [20–27].

2.5. Modelling Concentration Dependence on pH. Sorption
isotherms were obtained by measuring the europium con-
centration as a function of pH, adjusting the solutions to the
desired pH value (1 to 13), which were left for 24 h at 45 rpm.
The pH was measured using a HACH SensION + potenti-
ometer combined with an Ag/AgCl electrode. Likewise, the
modelling of the concentration dependence as a function of
pH considered the adjustment of three parameters: the
europium concentration, pH, and a factor called \( \pi \).

2.6. Thermodynamic Study. The distribution coefficients
\( (K_d) \) for tausonite and europium \( (5 \times 10^{-4} M) \) in a 0.5 M
KNO3 solution were measured at 283, 298, 308, 318, and
333 K using the previously described procedure [15]. All
experimental data are the average of duplicate batches with
good reproducibility (standard deviations were typically less
than 5% of the mean values).

Enthalpy change \( (\Delta H^\circ) \), entropy change \( (\Delta S^\circ) \), and Gibbs
free energy changes \( (\Delta G^\circ) \) at different temperatures were
estimated with equation (3) and europium sorption data
[28]:

\[
\Delta G^\circ = -RT \ln K_d^\circ T,
\]

where \( K_d^\circ \) is an equilibrium constant at absolute tem-
perature \( T \) and the perfect gas constant \( R = 8.3144721 \) J/kmol.

The two main thermodynamic parameters, such as \( \Delta H^\circ \)
and \( \Delta S^\circ \), can be determined from the basic thermodynamic
expression (equation (4)), where the Gibbs free energy is
expressed as follows:

\[
\Delta G^\circ = \Delta H^\circ - T \Delta S^\circ.
\]

To determine the two thermodynamic parameters \( \Delta H^\circ \)
and \( \Delta S^\circ \), the linearized equation of (4) was used. The line-
arized expression relates the Gibbs free energy at tempera-
ture \( (\Delta G^\circ/T) \) with the reciprocal of the absolute
temperature \( (1/T) \). Practically, all experiments fit the ther-
modynamic behavior of the experimental data by linear
regression (equation (5)) to determine the global thermal
character of the studied process (i.e., endothermic or
inverse) [28,29]:

\[
\frac{\Delta G^\circ}{T} = \Delta H^\circ \left( \frac{1}{T} \right) - \Delta S^\circ.
\]

3. Results and Discussion
3.1. Gamma Irradiation and Effect of Temperature on
Tausonite. The results obtained from the gamma irradiation
of tausonite showed the physical darkening of all the samples
because of the gamma dose received. On the other hand,
the diffraction patterns of the irradiated samples were compared
with the standard number card 35–0734. Figure 1 shows the
main diffraction peaks of the samples at 2θ = 23.0°, 32.4°,
40.4°, 57.8°, 67.8°, 77.2°, and 79.9° belonged to 100, 110, 111,
200, 211, 220, and 310, which correspond to tausonite. The
diffraction patterns show that in all cases, the material
presents an intensity of high diffraction peaks, which in-
dicates a high crystallinity. Furthermore, no phase changes
shifts or significant changes in diffraction patterns are ob-
served. However, a decrease in the intensity of the diffraction
peaks was observed. The changes observed in the intensity
lines can be attributed to two possible reasons. First, due to a
mutual torsion of tetrahedra under the effect of gamma
irradiation, the above could cause deformations within the unit cell due to a change in the distance between oxygen, iron, magnesium, or any metal ions [30]. The second reason is because the intensity lines can change in the diffractograms and can be explained by the supply of additional energy to the crystal influenced by gamma radiation, which can increase the vibrations of the crystal lattice [30]. Any of the above reasons can change the number of diffraction particles in the plane and thus change the intensity line in the diffractograms.

The micrographs of the irradiated samples are presented in Figure 2. The figure shows that the material has a morphology with aggregates of various polyhedral shapes tending to spherical shapes ranging in size from 500 nm to 1 μm. These results suggest that gamma irradiation causes agglomeration and reduction in particle size. Furthermore, the results of the BET analysis indicate that the samples at 12000 kGy have a mean pore diameter of 23.9 nm and a total pore volume of 0.087 cm³/g. Nonirradiated tausonite has a mean pore diameter of 13.6 nm and a total pore volume of 0.037 cm³/g. The specific area does not show significant changes between the irradiated and nonirradiated samples. These results suggest that gamma irradiation causes a slight increase in the porosity of the material.

Figure 3 shows the thermogravimetric curve of the tausonite sample. An endothermic peak is observed at 1009.73 K (736.58°C) that is related to the desorption of the water molecules added in the structure of the material; the above is verified with the gradual loss of the original mass of the sample (1.41%) [31, 32]. An exothermic peak occurs at 1168.12 K (894.97°C), with a rapid loss of mass equivalent to 1.88%. This weight loss has been observed by other authors and is usually associated with decomposition of SrCO₃ into SrO and CO₂ with the consequent release of CO₂ [33, 34]. This could be related to impurities from the tausonite synthesis process. By relating the XRD and TGA results, it can be established that tausonite is thermostable at the tested temperatures and can easily withstand the maximum thermal energy limit (2 kW/m²) recommended in the Waste Acceptance Criteria for disposal [18].

Figure 4 shows the IR spectra of the uncalcined tausonite samples and those calcined at 1023.15 K (750°C) and 1173.15 K (900°C). The figure shows the IR spectrum of uncalcined tausonite, where it can be observed that the absorption peaks at 3448 cm⁻¹ and 1633 cm⁻¹ are produced by the vibration of hydroxyl [31]. The hydroxyl group was probably derived from the adsorption water on the surface of the sample. At 2364 cm⁻¹, the presence of CO₂ (from the environment) absorbed by the sample was observed. The absorption bands at 1383 cm⁻¹ and 1458 cm⁻¹ are attributed to two tension modes of the unidentate CO₃ of SrCO₃ [35]. The broad absorption bands around 409 and 596 cm⁻¹ correspond to bending and stretching vibrations of the TiO₆ octahedron, respectively. The absorption peak at 596 cm⁻¹ belonged to the stretching vibration characteristic of the Sr-Ti-O bond [31, 36]. The overlap of the bands in the range of 471–523 cm⁻¹ of the infrared assignments shows the vibration of metal cations, such as Sr²⁺ [37]. At the tested temperatures, the absorption peaks at 3448 cm⁻¹ and 1633 cm⁻¹ caused by the hydroxyl vibration decrease as the temperature increases. Likewise, it is observed that the band at 1458 cm⁻¹ of CO₃ remains, which is explained by the presence of SrCO₃ (tausonite impurity).

3.2. Sorption Experiments

3.2.1. Sorption Kinetics Modelling. Figure 5(a) presents the results of europium sorption as function of time. These results were used to evaluate sorption kinetics and determine the retention capacity of tausonite. In Figures 5(b) and 5(c), the results of the linearization tests are presented to determine the order of the sorption kinetics using the Elovich (equation (6)) and pseudo-first-order equations (equation (7)):

\[
q_t = \alpha + \frac{2.303\beta}{\ln(\alpha)} \ln(t),
\]

\[
\ln(q_t - q_e) = \ln(q_e) - k_1 t.
\]

The fitting equations of expressions 6 and 7 were obtained by plotting \( q_t \) versus log \( t \) and \( \ln(q_t - q_e) \) versus \( t \), respectively. In both cases, the slopes and ordinates of the fitting equations allowed calculating the kinetic parameters of each of the models. As shown in Figure 5(b), the Elovich model exhibits a Pearson correlation and a statistical correlation coefficient \( (R^2) \) of 0.98 and 0.95, respectively. These correlations and confidence bands (95%) show that the model can mathematically explain the experimental data. The kinetic parameters \( \alpha \) and \( \beta \) of the model were calculated...
using the adjustment equation, obtaining $\alpha < 0$ and $\beta = 0.138$ mg/g, and the product of $\alpha \beta = 0$. Despite the fact that the adjustment presents high correlation values, it should be considered that the model restricts that the product of $\alpha \beta$ must be much greater than 1 [23]. In this context, it can be established that Elovich’s expression does not explain phenomenologically the europium retention process in tausonite.

Figure 5(c) shows the fit of the experimental data using the pseudo-first-order kinetic model (Laguerren equation).
As shown in the figure, the fitting equation shows a high correlation of Pearson and $R^2$ (0.98 and 0.94, respectively) with a 95% confidence level between the experimental data and the pseudo-first-order model. The slope and ordinate of the fitting equation allowed estimating the values of $k_1 = 0.0034 \text{ min}^{-1}$ and $q_e = 0.64 \text{ mg/g}$. The relative error ($E_R$) observed with respect to $q_e$ was estimated at 19.85%, which indicates that the model is relatively appropriate to explain the sorption of europium in the tausonite.

On the other hand, Figure 6(a) presents the results of the fitting of the experimental data with respect to the pseudo-second-order kinetic model given by

$$\frac{t}{q_t} = \frac{1}{h} + \frac{t}{q_e}$$

This figure was made by plotting $t/q_t$ versus $t$. The fitting equation obtained allowed estimating the kinetic constants $k_2 = 0.0026 \text{ g mg/min}$ and $q_e = 0.61 \text{ mg/g}$, with a Pearson correlation and $R^2$ of 0.99. The correlation coefficients and confidence bands show that the model satisfactorily explains the experimental data. Likewise, the magnitudes of $k_2$ and $q_e$ are congruent from a physical point of view, observing an $E_R$ of 14.23% with respect to $q_e$.

Comparing the pseudo-first-order and pseudo-second-order models, it is evident that both kinetic models present...
high correlation coefficients with a difference between the models of around 5.62% of $E_R$. It is observed that this discrepancy may be due to the number of experimental data points and some experimental fluctuations in dispersion around the general trend. The doubt about the choice between the first and second pseudo-order allows us to think of a process with a probable optimal pseudo-order ($k_n; n \neq 1$) to correlate the retention of europium in tautsonite:

$$q_t = q_e \left[1 - \left((n - 1)k_nt + 1\right)^{1/(1-n)}\right],$$

where the intermediate pseudo-order can be determined by nonlinear regression or by linearization with equation (10). As shown in Figure 6(b), the fitting equation allowed us to obtain the kinetic order of the europium retention process, which was estimated at a value of $n = 0.775$, $k_{0.775} = 0.0019$ min$^{-1}$, and $q_e = 0.97$ mg/g. The results of the statistical analysis exhibited a Pearson correlation and an $R^2$ of 0.98 and 0.97, respectively. Although the statistical analysis showed that the model is adequate to explain the retention process, the relative error associated with the sorption capacity in the equilibrium estimated by the model was very high ($E_R = 81.65\%$). Under these conditions, it is shown that this model does not physically explain the europium retention process in perovskites:

$$\frac{1}{(q_e - q_t)^{n-1}} = \frac{(n - 1)k_n t}{q_e^{n-1}} + \frac{1}{q_e^{n-1}}.$$

Additionally, the role of diffusive phenomena in the europium interaction process with an engineering barrier formed by tautsonite was studied. For this, the empirical model of fractional order was used, as expressed in

**Figure 6:** (a) Fitting of experimental data using the pseudo-second-order model. (b) Fitting of experimental data using pseudo-order model ($k_n; n \neq 1$). (c) Fitting of experimental data using intraparticle model.
\[ q_t = k_{0.5} \sqrt{t} + C. \] (11)

This model allows defining the influence of intraparticle diffusion on the solute retention process [38]. In this context, Figure 6(c) represents \( q_t \) as a function of \( t^{0.5} \). The fitting equation presents an adequate Pearson correlation and \( R^2 \) of 0.94 and 0.87, respectively. This fitting equation allowed estimating the kinetic parameters of the model: \( k_{0.5} = 0.0119 \text{mg/g.min}^{0.5} \) and \( C = 0.0545 \text{mg/g} \), where \( C \) is related to the thickness of the surface layer formed on the tausonite [23]. However, to give a possible additional physical significance to the intraparticle diffusion rate constant \( (k_{0.5}) \), equation (11) was rewritten in a new expression (see equation (12)) with a new derived parameter \( \tau \), which designates the characteristic intradiffusion time and the normalized intraparticle diffusion rate. In this case, \( \tau \) was estimated at 7.06 min:

\[ q_t = \frac{k_0}{k_{id}} q_e \sqrt{t} + C, \] (12)

where

\[ \tau = \frac{1}{k_{id}^{0.5}}. \] (13)

The results of fitting of the experimental data suggest that the europium retention process in taursonite can be explained by interpreting three zones [39], which can be seen in Figure 6(c). Zone I is attributed to the diffusional process of EuNO\(_2\)\(^+\) on the surface of the material; therefore, this was the fastest sorption period. Zone II suggests a slight intraparticle diffusion, which occurs in a delayed process. Zone III could be considered as the diffusion through the smallest pores of the material, which is followed by the establishment of equilibrium. This behavior could be related to the increase in porosity caused by irradiation of tausonite. This statement is supported by observations made by other authors, who have indicated that when the fit data respond to more than one line of fit, it is indicative that the sorption process is influenced by more than one diffusion phenomenon [13, 40, 41].

The results of fitting of the experimental data with the kinetic models studied clearly show that the sorption of europium in taursonite follows a pseudo-second-order behavior, which implies that the europium is retained in the taursonite by the formation of strong bonds. Different authors have observed that when the experimental data are fitted to a pseudo-second-order model, they indicate the formation of strong bonds and, in some cases, chemisorption between sorbents/sorbates [20, 24]. In this context, the observed retention process is consistent with previous studies in taursonite, which have demonstrated that the material forms bidentate surface complexes associated with the interaction of the surface sites TiOH\(_2\)+ and (TiOH)\(_2\) [14]. Furthermore, the results obtained by fitting the intraparticle model indicate that the sorption mechanism is influenced by more than one diffusion phenomenon, and they contribute to the formation of surface complexes. This diffusive process is physically related to the parameter \( \tau \) estimated in the model (\( \tau = 7.06 \text{ min} \)), as has been observed with other materials [39–41].

From a practical point of view, an engineering barrier is required to have physical and chemical properties that provide containment of radionuclides and prevent them from being released and migrating from a waste package [16]. The characteristics of the waste package are defined by the waste acceptance criteria; thus, the waste package represents the first containment barrier for radionuclides and usually consists of a waste container that is filled with a solid matrix (encapsulant) that is sealed with a lid [17, 18]. The most important safety criteria of the waste package during the operational period and after the closure of the disposal site are the chemical and physical properties of the waste package, the dimensions and weight of the package, permissible levels of radioactivity, radiation dose rate, and allowed heat generation [16–18].

In order to carry out a preliminary evaluation of taursonite as an engineering barrier in a normal scenario of safety assessment, two basic aspects of the waste package should be considered: the permissible specific activity of long-lived alpha-emitting radionuclide (4000 Bq/g) and a thermal power of approximately 2 kW/m\(^3\) [16–18]. In this context, in an extreme hypothetical case in which all alpha-emitting radionuclides (Ac\(^{3+}\)) were released from the waste package, an engineered taursonite barrier would have to retain at least 26.59 g of alpha-emitting radionuclides, equivalent to 4000 Bq/g of actinides (Ac\(^{3+}\)) isotopes and a waste package of approximately 900 kg. In this scenario, if the taursonite has a sorption capacity of 0.54 mg of radionuclides per gram of taursonite, approximately 50 kg of engineered barrier would be needed as a buffer material, which is an acceptable value from a technical and economic point of view. Considering that a thermal power of 2 kW/m\(^3\) is much lower than the thermal processes tested, it is possible to affirm that the taursonite would not suffer deterioration due to thermal fields and gamma irradiation, as has been demonstrated in the experimental tests.

3.2.2. Modelling Concentration Dependence as a Function of pH. Figure 7 shows the concentration dependence as a function of pH. The two plateaus at the beginning and end of the pH range suggest that equation (14) has only three adjustable parameters, \( q_t \), pH\(_0\), and \( \pi \) to fit this variation:

\[ q_t = \frac{q_e (pH-pH_0/\pi)}{1 + e^{(pH-pH_0/\pi)}}, \] (14)

where pH\(_0\) = 6.2 and \( \pi = 2.2 \).

Considering that the error of the experimental measurements is less than 5%, Figure 7 shows a reliable concordance between the experimental and estimated values. As can be seen in the figure and in equation (14), pH\(_0\) is the value of the jump center where the sorption capacity of taursonite reaches half the limit retention capacity, \( q_e \). The jump range (\( \Delta \text{pH} \)) is approximately

\[ \Delta \text{pH} = \text{pH}_0 \pm \frac{\pi}{2}. \] (15)

Furthermore, it is probable that pH\(_0\) is in causal correlation with the diagram of Figure 7. It should be noted that
the speed of the jump is characterized by the amount of π = 2.2 (see equation (16)).

\[
pH = pH_0 + \pi \ln \frac{q_{hl}}{q_e - q_t},
\]

where π is also physically equivalent to the amount of pH. In this context, it is possible that its value (2.2) is related to any optimal property value, or the beginning of a certain phenomenon, equilibrium phase, or the formation or disappearance of a certain species. Previous studies of the variation of europium concentration versus pH in taursonite showed that this curve can be deconvoluted and explained by the formation of a bidentate surface complex ((TiOH)2EuNO23+; log K = 9.08 ± 0.02), the chemical precipitation, and surface europium hydroxyl and carbonate species [15]. The pH of 6.2 associated with the physical parameter π coincides with the saturation of the surface sites of the species TiOH23+ with EuNO23+ that form the bidentate complex, at the beginning of the chemical and surface precipitation of the hydroxylated and carboxylated species of europium and which validates the hypothesis of the physical meaning of the parameter π deduced with the proposed model. Due to the mathematical property of the suggested model, it is likely that 2π = 4.4 has the physical significance that explains the equilibrium between the surface complexes that taursonite forms on europium. It is important to note that this equation must be handled with great mathematical caution due to the definition domain (0° < q_t < q_e).

Based on the above, it is possible to establish that this model can be used to predict values of sorption capacity for pH other than the experimental data points. Furthermore, the parameter values can be an excellent criterion when comparing them with the values of other similar studied systems under different conditions.

3.3. Thermodynamic Study. The values of the enthalpy change (ΔH°), entropy change (ΔS°), and Gibbs free energy change (ΔG°) at different temperatures were estimated with equation (3) and data on the europium sorption data. In Figure 8(a), the variation of ΔG°/T is presented as a function of the inverse of the reciprocal temperature (1/T) for the studied system, obtained with equation (5) (linear regression) and equation (17) (second-grade nonlinear regression). A preliminary inspection of the systems studied through Figure 8(a) shows a discrepancy between the straight line and the uncertainty bars of some of the experimental scattering points. Therefore, it can be established that the nonlinear regression fitting is the best to reduce this discrepancy and obtain more accurate and true values of the two thermodynamic parameters ΔH° and ΔS°. So, equation (5) is replaced by a polynomial equation with two or three degrees expressed as follows [28]:

\[
\Delta G^°_T = A_0 + A_1 \left( \frac{1}{T} \right) + A_2 \left( \frac{1}{T} \right)^2 + A_3 \left( \frac{1}{T} \right)^3 + \ldots .
\]

It can be seen that the mathematical derivation of equation (4) allows determining ΔH° and ΔS° using the following equations [42]:

\[
\Delta H^°(T) = \left( \frac{\partial (\Delta G/T)}{\partial (1/T)} \right)_p,
\]

\[
\Delta S^°(T) = - \left( \frac{\partial (\Delta G/T)}{\partial (T)} \right)_p = \frac{\Delta H^° - \Delta G^°}{T}.
\]

In Figure 8(b), the variation of enthalpy as a function of temperature is shown, obtained by nonlinear regression with equation (18). In this figure, the horizontal dashed line represents the constant value determined by linear regression. From these figures, it can be determined that the values of thermodynamic parameters, ΔH° and ΔS°, obtained by nonlinear regression can vary around and on both sides of the constant values obtained by linear regression. For this reason, several interpretations made by experimenters in the literature using only linear regression give a general description and are globally reliable but cannot give precise temperature effects that allow choosing which optimal temperature is preferable.

In this study, ΔG°, ΔH°, and ΔS° are negative; additional information indicates that this negativity is accentuated when the temperature increases, which means that the process releases more energy with the temperature. The magnitude of the thermodynamic potential makes it possible to establish that europium retention in taursonite is an exothermic and spontaneous reaction at the tested temperatures, and these observations are consistent with sorption thermodynamics when the surface of the material is not yet saturated [28]. On the other hand, the magnitude of ΔH° indicates that the sorption of europium is due to the formation of surface complexes. Studies using time-resolved
laser-induced fluorescence spectroscopy and X-ray photoelectron spectroscopy show that the energy transfer between the Eu³⁺ ions sorbed on the tauronite follows a dipole-dipole interaction and the formation of a bidentate inner-sphere surface complex [43]. Other complexes on the surface of the inner sphere have been observed in the retention of europium in various solids [19, 44]. In the same context, \( \Delta S^\circ \) is globally negative and its negativity is accentuated when the temperature increases; this tendency of the \( \Delta S^\circ \) is explained by the imbalance of the surface forces in the material caused by the transfer of sorbate from the aqueous phase to the surface of the sorbent [28].

However, there is an interesting particularity; the entropy is cancelled at \( T = 288.65 \text{ K} \) and takes positive sign for lower values than 288.65 K. The positive value found at this temperature may indicate an increase in the disorder in the liquid/solid interface during the interaction of the surface sites with EuNO₂⁺; another reason could be due to the greater number of ions that leave the surface of the solid when they are exchanged for EuNO₂⁺, particularly if the
charge density of EuNO$_2^{+}$ exceeds the charge of the ions exchanged outside the surface of the tausonite [29]. On the other hand, the fact that $\Delta S^* = 0$ at $T = 288.65$ K makes it possible to propose eliminating the temperature as an independent variable and plot in Figure 8(a) $\Delta H^*$ versus the entropy $\Delta S$. Then, the practically linear enthalpy-entropy dependence shown in Figure 8(d) can be expressed as follows:

$$\Delta H^* = T_c [\Delta S^* (T) - \Delta S^*], \quad (20)$$

where $T_c$ and $\Delta S^*$ are the characteristic temperature and the characteristic entropy of the studied system, respectively. By analogy, if we can set that also $T_c$ $\Delta S^*$ product is equivalent to the characteristic, $\Delta S^*$ can be expressed as follows:

$$\Delta H^* = T_c [\Delta S^* (T) - \Delta S^*]. \quad (21)$$

And equation (21) can be rewritten as follows:

$$\Delta H^* = T_c [\Delta S^* (T) + \Delta S^*]. \quad (22)$$

Then, it follows that the investigated system has two main independent specific parameters, $\Delta H^*$ and $\Delta S^*$, and a dependent parameter $T_c$, which can be derived simply by means of equation (21). The estimated values of the characteristic individual thermodynamic parameters for the studied system were $\Delta H^* = -11.326$ kJ/mol and $\Delta S^* = 36.684$ J/kmol with $R^2$ of 0.99. These values correspond to a $T_c = 35.6 \, ^\circ C$ (308.75 K). However, $\Delta H^*$ and $\Delta S^*$ can be considered as independent and characteristics specific parameters for a well-determined and investigated system, while in the case of $T_c$, care must be taken because it cannot be characteristic of the studied system; it is only an intermediate mathematical variable that obeys equation (23) by simply indicating approximately the temperature of the system during its processing:

$$\frac{\partial \Delta H^*}{\partial T} = \frac{C_p}{(C_p/T)} = T. \quad (23)$$

Additionally, it can be observed that $T_c$ is very close to the value of the mean temperature ($T_{md} = 37.5 \, ^\circ C$) of the studied system. This mathematical property is due to the relationship between the slope and the intercept on the ordinate of the tangent line to the curve at a given point for the function $\Delta G^* / T = f (1 / T)$ when this point moves along the curve line (Figure 8(a)), that is, when $T$ varies in the working temperature range. When a thermodynamic parameter $\Delta H^* (T)$ or $\Delta S^* (T)$ takes the identical value to one of its corresponding specific parameters $\Delta H^*_c$ or $\Delta S^*_c$, the other thermodynamic parameter becomes null, and the thermal character of the process takes on a reciprocal behavior.

Finally, this finding allows carrying out an interesting comparative study in the future to find an interpretation of the values, amount, and physical significance of the true characteristic parameters $\Delta H^*_c$ and $\Delta S^*_c$ for each system studied in relation to its properties and the experimental conditions of the process.

4. Conclusions

The results obtained from the study of the effects of gamma irradiation on tausonite allowed establishing that at the irradiation doses tested, the material undergoes some morphological changes, such as darkening, agglomeration, reduction in particle size, and a slight increase in porosity. Evidence indicates that irradiation does not cause appreciable damage to the crystal structure and does not produce phase changes. The results on the effects of temperature indicate that the material is thermostable at the tested temperatures. Thus, with the evidence presented, it can be stated that the perovskite studied would not suffer deterioration due to thermal fields and gamma irradiation, which could be expected in a radioactive waste disposal site.

On the other hand, the thermodynamic and sorption studies showed that the sorption of europium on tausonite occurs through an exothermic and spontaneous reaction at the tested temperatures and that it follows a pseudo-second-order behavior. This implies that europium is retained in tausonite by the formation of bidentate inner-sphere surface complexes, where Eu$^{3+}$ ions bind to sites on the tausonite surface by dipole-dipole interaction, as shown by the thermodynamics of the parameters and the different sorption models tested. The evidence obtained suggests that diffusion of europium at the liquid/solid interface contributes to the initial process of interaction between europium and sites on the surface of the material. This phenomenon could be related to the increase in porosity caused by the irradiation of the material. The modelling of the sorption with respect to the proposed pH allowed determining the physical parameter $\pi$. The evidence obtained suggests that $\pi$ is a physical parameter that relates pH to an optimal property value, for example, the equilibrium phase, the formation, or disappearance of chemical species, among others. In this study, due to the mathematical property of the proposed model, it is likely that $2\pi = 4.4$ has the physical meaning that explains the equilibrium between the surface complexes that tausonite forms with europium. Therefore, it can be established that this model can be used for prediction of sorption capacity values for pH other than the experimental data points. In addition, the values of $\pi$ parameters can be an excellent criterion, when compared with the values of other similar systems studied under different conditions.

Likewise, the evidence suggests that 50 kg of tausonite would have the capacity to retain at least 26.59 g of alpha-emitting radionuclides (Ac$^{3+}$), equivalent to a waste package (900 kg) with a maximum activity of 4000 Bq/g. This is quite an acceptable value from a technical and economic point of view.

Finally, the evidence obtained from the evaluation of tausonite, such as its resistance to gamma irradiation and temperature, sorption capacity, and thermodynamic behavior, suggests that this perovskite could be a potential candidate for its use in an engineering barrier in the disposal of radioactive waste. However, additional studies are required on the effects of neutrons on the structure and morphology of tausonite.
Data Availability

The datasets used and/or analysed during the current study are available from the corresponding author on reasonable request.

Disclosure

This work is part of the DR-001 project of the Radioactive Waste Department, National Institute of Nuclear Research (ININ). This work was supported by Benemerita Autonomous University of Puebla. The irradiation of the material was performed by the staff of the ININ gamma irradiator. Dose calculations were estimated by Ing. Hector Carrasco Abrego.

Conflicts of Interest

The authors declare that they have no conflicts of interest.

Authors’ Contributions

Ortiz-Oliveros H.B. was responsible for conceptualization, investigation, supervision, data curation, methodology, formal analysis, and chemical analysis, prepared the original draft, and reviewed and edited the manuscript. Flores-Espinosa R.M. reviewed and edited the manuscript and performed data validation and chemical analysis. Nour-eddine Ouerfelli was responsible for thermodynamic modelling and mathematical analysis. Avila-Perez P. and Cruz-Gonzalez D. performed data validation and reviewed and edited the manuscript.

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