An effective utilization of MXene and its effect on electromagnetic interference shielding: flexible, free-standing and thermally conductive composite from MXene–PAT–poly(p-aminophenol)–polyaniline co-polymer†

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MXene and conductive polymers are attractive candidates for electromagnetic interference shielding (EMI) applications. The MXene–PAT-conductive polymer (CP) composites were fabricated by a cost-effective spray coating technique and characterized using X-ray photoelectron spectroscopy (XPS), scanning electron microscopy (SEM), X-ray diffraction (XRD) and Raman spectroscopy. A new approach has been developed for the synthesis of exfoliated MXene. The MXene–PAT–poly(p-aminophenol)–polyaniline co-polymer composite exhibited good electric conductivity (EC) of 7.813 S cm−1. The composites revealed an excellent thermal properties, which were 0.687 W (m·K)−1 thermal conductivity, 2.247 J (g·K)−1 heat capacity, 0.282 mm2·s−1 thermal diffusivity and 1.330 W m−1·K−1 thermal effusivity. The composites showed 99.99% shielding efficiency and the MXene–PAT–PANI–PPA composite (MXPATPA) had EMI shielding effectiveness of 45.18 dB at 8.2 GHz. The reduced form of MXene (r-Ti3C2Tx) increased the shielding effectiveness (SE) by 7.26% and the absorption (SEA) was greatly enhanced by the ant farm like structure. The composites possess excellent thermal and EMI SE characteristics, thus can be applied in areas, such as mobile phones, military utensils, heat-emitting electronic devices, automobiles and radars.

1 Introduction

Widespread electromagnetic interference (EMI) is a demanding issue as a consequence of the rapid advancement of miniature electronic components and extensive usage of wireless communication worldwide. The global growth in human population together with the mass production of a variety of electronic devices is creating electromagnetic pollution. EMI is a hot topic, due to the effect on the proximal biological and electronic systems.1,2 In general, the passage of migrant birds dramatically changes from year to year, due to electromagnetic noise (EMN), which disturbs the natural compass and magnetic positioning capability of migratory birds. Hence, EMN restricts seasonal and natural habitant birds in rural, town, and wild areas.1,3 Furthermore, EMI also causes various health issues in human, such as headaches, chromosomal mutation, and nervous disorders.3 When electromagnetic radiation hits a living organism, a dipole environment is created in the body, which creates additional potential, and diminishes the ions transportation through nerves. EMI induces vibration of water molecules, which releases extra heat energy, which is disseminated among organs to generate physiological dysfunction, unhealthy infant, denature of protein, disorder in enzyme function, non-regular function of organelles, and reduction of
fertility, which raise questions about the sustainability of human and other living things on the Earth.\textsuperscript{5,7}

EMI shielding is expressed in dB, and is inevitable in the modern electronic world.\textsuperscript{2} Non-Ionized Radiation (NIR) has been widely used, as it is safe, even though recent study reports that some NIR radiation has a detrimental issue on biotic structure. The NIR includes extremely low frequency, radio-frequency, microwaves, visible light, and some part of ultraviolet (UV), which can be categorized into perceptible and non-perceptible radiation. The visible region is harmful, whereas some frequencies of imperceptible radiation potentially have considerable influence on human health. The electron moving materials are the primary source of NIR, while the surrounding environment is a scattering source of NIR.\textsuperscript{6-8} Thus, achieving electromagnetic compatibility is indispensable in the current norm. Electromagnetic noise is engendered by various sources, such as thermal background noise, EMI, and terrestrial noise. Diminishing or blocking electromagnetic noise without disturbing adjacent components is considered to be electromagnetic compatibility.\textsuperscript{9,10}

To accomplish good electromagnetic compatibility, scientists worldwide utilize numerous strategies by using various materials, such as MXene, metal plate, textile, wax, nonconductive polymer matrix, carbon fabric, intrinsic conductive polymer (ICP), graphite, graphene, carbon nanotube, nanoplate, nanoparticle, plant materials, carbon black, and cement.\textsuperscript{11-18} Nonconductive polymers, such as polystyrene, polypropylene, polyethylene, and polystyrene\textsuperscript{21} are used as supportive matrices for EMI shielding applications. ICPs, such as polyaniline (PANI),\textsuperscript{19} polypropylene (PP),\textsuperscript{22} and polystyrene\textsuperscript{23} are used as a single matrix, or in combination. The combination of different materials is used to create different structural features with enhanced EMI shielding.\textsuperscript{17-28} Absorption, reflection, and multiple reflections are central factors for the EMI shielding behaviour of materials, where one aspect of the basic mechanism is more dominant than others. Absorption can be enhanced by increasing the bipolarity of the composite, whereas reflection depends on the electric conductivity of the materials. Absorption happens due to scattering, multiple reflections, and electric and magnetic bipolarity. Nanoparticles possess strong scattering ability, and are able to absorb different energy electromagnetic radiation. Multiple reflections can be enhanced by layered structure and electron flow. The dipolar environment can be created by using polar fillers, and hollow and triangular nanomaterials. Uneven electron distribution in hollow and triangular nanostructure creates polar components that affect EMI shielding.\textsuperscript{19-31,29,30}

\begin{table}[h]
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\hline
Material & EMI shielding & SSE \\
\hline
Ti\textsubscript{3}C\textsubscript{2}T\textsubscript{x} & 74.9 dB & 318 dB g\textsuperscript{-1} \\
\hline
Mxene/polypyrrole (PPy) textile & 47 dB & 0.2 cm thickness \\
\hline
PEDOT : PSS/waterborne polyurethane (WPU) composite & 62 dB & 0.2 cm thickness, 318 dB g\textsuperscript{-1} \\
\hline
\end{tabular}
\caption{EMI shielding and SSE of different materials.}
\end{table}

Ti\textsubscript{4}C\textsubscript{2}T\textsubscript{x}/co-doped polyaniline composite exhibits 36 dB of EMI SE with 24.4 S cm\textsuperscript{-1}. The co-doped shows 16 dB of EMI SE and 0.3 S cm\textsuperscript{-1} of EC while pure polyaniline (PANI) exhibits 20 dB of EMI SE and 0.213 S cm\textsuperscript{-1} of EC. In addition, carbon nanotube/polyaniline films displays the EMI shielding of 74.9 dB with 7.5 \times 10\textsuperscript{7} dB cm\textsuperscript{-1} g\textsuperscript{-1} of SSE, 3009 S cm\textsuperscript{-1} of EC and thickness of 0.029 mm. The MWNT/PANI (Ku band), SWCNT/PANI (2–18 GHz), MWNT/PANI (0.8–3 GHz) composite show 39.2, 31.5, 47 dB of EMI shielding and its corresponding thickness are 2, 4 and 0.5 mm, respectively. PEDOT : PSS/waterborne polyurethane (WPU) composite exhibits EMI shielding of 62 dB with 0.15 mm.\textsuperscript{24-28} Therefore, increasing thickness of the conductive polymer composites reduces the EMI shielding. Hence, conductive polymer composition should be a thin film.

MXene is a novel kind of two-dimensional ternary metal carbide/nitride derived from the corresponding three-dimensional MAX phase. The MXene possess excellent physical properties such as hydrophilic in nature with good electrical and thermal conductivity, higher specific surface area and excellent film forming ability which are main reason for the preference of MXene than graphene. Hence, MXene and MXene composites become hot topic for EMI shielding application around the world. The various type of MXene have been studied and EMI shielding application mainly focus on Ti\textsubscript{3}C\textsubscript{2}T\textsubscript{x}. The Shahzad et al. reported that the Ti\textsubscript{3}C\textsubscript{2}T\textsubscript{x}/sodium alginate coating with 45 µm on Polyethylene terephthalate (PET) film shows 92 dB of EMI SE with 4600 S cm\textsuperscript{-1} of electric conductivity.\textsuperscript{14,27-28} The ultralight MXene-based aerogels exhibits 75 dB EMI shielding with SSE of 9904 dB g\textsuperscript{-1} cm\textsuperscript{3} and density of 0.004 g cm\textsuperscript{-3} where aerogel induced by freeze drying process while MXene Foam exhibits 70 dB of EMI SE with 0.006 cm of thickness 0.22 g cm\textsuperscript{-1} density and SSE of 318 dB g\textsuperscript{-1} cm\textsuperscript{3} where foaming induce by hydrazine.\textsuperscript{16-24} Moreover, MXene/calcium alginate aerogel exhibits 54.3 dB of EMI shielding with SSE of 40.22 dB g\textsuperscript{-1} cm\textsuperscript{3}, thickness 26 µm of and density of 1.35 g cm\textsuperscript{-3}.\textsuperscript{25} Hence, introduction of polymers comparatively reduces the EMI shielding of pure MXene, increase the density and give rise various structural features. The graphene/copper nanowire embedded graphene aerogel exhibits 47 dB of EMI SE with 0.2 cm of thickness and EC of 1.208 S cm\textsuperscript{-1}.\textsuperscript{26} The MXene polypyrrole (PPy) textile with thickness 1.3 mm (3 times coated) displays EMI SE of 90 dB with 10 S cm\textsuperscript{-1} of EC whereas Ti\textsubscript{3}C\textsubscript{2}T\textsubscript{x}/poly(3,4-ethylenedioxythiophene)–poly(styrenesulfonate) (PEDOT : PSS) (7 : 1) with 11.1 µm thickness and 340.5 S cm\textsuperscript{-1} of EC shows 42.1 dB of EMI SE.\textsuperscript{32-33} Hence, the MXene is a good choice for EMI shielding compared to the other fillers which is due to the MXene have low density, high electric conductivity, high EMI shielding with micron thickness.

The selective etching strategy has been used to synthesize MXene, for which HF, LiF/HCl, FeF\textsubscript{3}/HCl, and NH\textsubscript{4}F/HCl are used. However, the minimally intensive layer delamination (MILD) method uses LiF/HCl, which gives rise to the best quality MXene, compared with other etchants.\textsuperscript{31} The general formula of MXene is M\textsubscript{n+1}X\textsubscript{n-1}T\textsubscript{x}, while the corresponding MAX phase possesses M\textsubscript{n}AX\textsubscript{n-1} universal symbolization, where M is an early transition metals, A is a group 13/14 element, X is carbon or nitrogen, and T\textsubscript{x} is a surface functional group, which include –OH, –F, –O and Cl. During the etching process, the A layer is eradicated, and the corresponding bond between M–A and A–X are substituted by the surface functional groups. Thus, the MXene is formed with the gap between 2 layers. One A layer is a sandwich between two M–X layers, where the M–X bond is stronger than the M–A bond. Hence, depletion of M–A bond is easier than of M–X bond, which remains as MXene.\textsuperscript{34-35} So far, many MAX phases and MXenes have been studied, even though Ti\textsubscript{3}AlC\textsubscript{2} and its corresponding MXene are widely used for various applications, such as EMI shielding, battery, medical, and capacitors.\textsuperscript{31-36}
In this study, we developed a new type of polymer composition called "PAT polymer" solution and a new approach has been made to prepare exfoliated MXene (d-MXene) and reduced MXene (r-MXene). According to our knowledge, this is the first time that the conductive polymers (CP), such as poly(p-aminophenol) (PPAP), polyaniline–PPAP (PANI–PPAP) copolymer were employed for EMI shielding application. We have prepared and analyzed the effectiveness of other CPs, such as polypyrrole (PPy), polyaniline (PANI), polythiophene (PTh) for this study. The general formula of composite was MXPATCP; where MX is MXene, PAT is polymer composition and CP is conductive polymer. Hence, MXPATPN, PXPATPA, MXPATPNPA, MXPATPpy and MXPATPTh composites were successfully fabricated with corresponding conductive polymers, which are PANI, PPAP, PANI–PPAP, PPy and PTh, respectively. The composites without CP are MXPAT, dMXPAT, uMXPAT and rMXPAT which were prepared by using MXene, d-MXene, u-MXene and r-MXene, respectively. The rMXPAT and dMXPAT were used to compare the EMI SE with other composites. The conductive carbon fiber network was denoted as CCNW and functionalized CCNW was denoted as FCCNW. The parameters such as elemental analysis, structural features, thermal stability, thermal conductivity, electric conductivity and EMI shielding were investigated in detail.

2 Materials and method

2.1. Materials

Titanium, Titanium carbide, and aluminium (325 mesh size) were purchased from Materials Korea (Rep. of Korea). Graphene (M-25, average size of 25 μm) was obtained from Ditto Technology (Rep. of Korea). Ethanol (98%), agar, hydrochloric acid (HCl 35%), tetrahydro furan (THF), and aniline were supplied from Samchun (Seoul, Korea). Polyethylene terephthalate (PET) binder (fibre diameter 2.2 dtex, 5 mm) and carbon fibre (fibre diameter 7 μm, 6 mm length) were obtained from TORAY product, (Tokyo, Japan), while polycrylamide (PAM) was purchased from Sigma Aldrich (Seoul, South Korea). Iron(II) chloride (FeCl3 97%), lithium fluoride (LiF 300 mesh), pyrrole (98%), thiophene (99%), trimesic acid (98%), 4-aminophenol (98%), acrylic acid, N-methyl-2-pyrrolidone (NMP), and polyacrylic acid (PAA) were obtained from Sigma Aldrich (Seoul, South Korea). Polyvinyl alcohol (PVA 89%) and polyvinylidene fluoride (PVDF) were supplied by Alfa Aesar (Seoul, Korea). Chitooligosaccharide (10 000 MW) was provided by Sunchon National University, Biomedical polymer lab, (Jeonju Korea). Glycerin was obtained by reagent from Duksan (Seoul Korea). N,N-Dimethylformamide (DMF) was purchased from Daejung (Seoul Korea). All of the chemicals were used without further purification.

2.2. Synthesis of Ti3AlC2

TiC, Ti, and Al powders with the molar ratio of 2 : 1 : 1 were ball-milled by using Pulverisette 6 Planetary Mono Mill (Fritsch, Germany) in ethanol medium at 200 rpm for 1 h under nitrogen environment, and the resultant mixture was dried at 80 °C for 12 h in vacuum oven. Then, 3 g of the above mixture was cast into disc of 12 mm diameter by applying 27.6 MPa pressure for 5 min in a laboratory press. Subsequently, the disc was treated at 1350 °C with a heating rate of 20 °C min⁻¹ in argon environment for 2 h, and cooled to room temperature (RT). The resultant disc was powdered by ball milling in ethanol medium at 300 rpm for 3 h under nitrogen gas. The ethanol was evaporated at 80 °C for 12 h in vacuum oven, and the resultant MAX phase was directly utilized for MXene synthesis.

2.3. MXene (Ti3C2T x) synthesis

First, 1 g of Ti3AlC2 and 1 g of LiF were mixed together in 20 mL of 9 M HCl solution, and stirred at 35 °C for 24 h. The etched product was washed with DI water up to pH 6 by centrifuging at 3500 rpm for 5 min. During the washing process, the colours of the supernatants shown in Fig. 1a and b differ from that of DI water, which is due to the dispersion of MXene flake. The obtained MXene was dehydrated in vacuum oven, until achieving constant weight. The resultant MXene was directly used for the colloidal solution, exfoliated MXene, and composite preparation.

2.3.1. MXene suspension solution and exfoliated MXene (d-Ti3C2T x) preparation. The etched MXene was directly transferred to DI water, and sonicated at (15–17) °C for 1 h under nitrogen environment. The resultant solution was then centrifuged at 3500 rpm for 30 min, and the supernatant was collected and stored at 5 °C for the further process, termed MXene suspension. One portion of MXene suspension was filtered through 0.45 μm nylon membrane filter, and dried at 80 °C in vacuum oven until constant weight was obtained. The concentration of the colloidal solution was calculated as 200 mg L⁻¹, and denoted as d-Ti3C2T x. Another approach that was taken was that the colloidal solution was evaporated at rotavapor at 60 °C, and the volume of colloidal solution reduced. The solution was cooled immediately by pouring cold water onto the evaporating flask, then nitrogen gas was pumped into the rotavapor, until it reached atmospheric pressure. Subsequently, the evaporating flask was removed, and cooled to 5 °C in 30 min. The cold mixture was filtered through 0.45 μm nylon membrane filter, and the solid was dried at 80 °C in vacuum oven, until constant weight was obtained, and denoted as d-Ti3C2T x. This exfoliation process was repeated several times.
times, until obtaining clear supernatant, and the resultant solid was dark brown in colour, and denoted as u-MXene. Both d-MXene and the u-MXene were collected for further process. The colloidal solution was prepared by using d-MXene by sonicating 30 min in aqueous phase for composite preparation (Scheme 1).

### 2.4. Conductive polymer (CP) synthesis

#### 2.4.1. Synthesis of polyaniline (PANI). Next, 13.67 mL of aniline was dissolved in 750 mL of 0.2 M HCl, and stirred for 15 min. Then, 0.25 M of ammonium persulfate was added drop-wise into aniline with continuous stirring, and kept 24 h for polymerization, without stirring. The polymer was collected at the filter, and washed thoroughly with an ample amount of 0.2 M HCl, followed by 750 mL of acetone. The product was dried at 60 °C in vacuum oven, and stored at RT for further process.\(^{36}\)

#### 2.4.2. Synthesis of poly \(para\)-aminophenol (PpAP). The particular amount of \(para\)-aminophenol (pAP) (0.2 M) was dissolved in 1000 mL of 0.5 M HCl, and stirred for 15 min; 500 mL of 0.6 M APS was added drop-wise, and stirred for 24 h at RT. Then, the reaction was terminated by adding an amount of methanol. The product was filtered, and washed by 0.5 M HCl, until clear filtrate was obtained, followed by washing with methanol, and then acetone. Each washing was performed until clear filtrate was obtained. The product was then dried at 80 °C for 12 h, and used for further process.\(^{37}\)

#### 2.4.3. Synthesis of the poly aniline–poly \(para\)-aminophenol copolymer (PpAP–PANI). Next, 0.1 M of aniline and 0.1 M of pAP were mixed in 1000 mL of 0.5 M HCl, and stirred for 15 min. Then 1000 mL of 0.3 M of APS was added drop-wise, and stirred for 24 h. The yield was filtered, and washed with 0.5 M HCl, methanol, and acetone [S1, S2]. The product was dried at 80 °C in vacuum oven, and stored for further process.\(^{36,37}\)

#### 2.4.4. Synthesis of poly-pyrrole (pPy). The 0.05 M of pyrrole, 0.1 M FeCl\(_3\), and 0.0225 M of SDS were dissolved and stirred at RT for 4 h. The resultant black colour product was collected at filtration, and washed with a sufficient amount of DI water and ethanol. The collected yield was dried at 40 °C for 12 h in vacuum oven.\(^{38}\)

#### 2.4.5. Synthesis of poly-thiophene (pTh). Then, 1 g of SDS, 2 g of CTAB, 21 mL of thiophene, and 0.25 g of CuCl\(_2\) were sonicated in 500 mL of DI water for 30 min. The 90 g of APS was dissolved in 500 mL DI water which was added drop-wise into thiophene solution while stirring for 2 h. The resultant mixture was then kept for 24 h, without being disturbed. After that, the yield was washed with ample amount of DI water and ethanol. The product was dried at 50 °C, until constant weight was obtained.\(^{39,40}\)

### 2.5. Preparation of conductive carbon fibre network (CCNW) and functionalization

Next, 600 g of carbon fiber, 150 g of PET binder, and 0.3 wt% of dispersant (PAM) were dispersed in a sufficient amount of deionized (DI) water at 500 rpm for 10 min. A drum dryer with 140 °C surface temperature and 7 m min\(^{-1}\) speed was used to produce CCFN. The areal density of the obtained CCFN was 30 g m\(^{-2}\). The CCNW with 29 cm \(\times\) 21 cm dimension was dipped into the 2 g L\(^{-1}\) chitooligosaccharide solution for 2 min, and dried at 100 °C for 12 h. The gained CCNW was denoted as FCCNW, and was used directly for composite preparation.

### 2.6. Preparation of the PAT solution

The PAT solution was prepared according to Table 1. All the compositions were mixed together, and stirred for 30 min (Fig. 2a), and then the temperature was gradually increased to 130 °C, and the mixture boiled, until it became transparent.
yellow (Fig. 2b). The continuous boiling process gave rise to a transparent brown solution (Fig. 2c), which was obtained as the final composition. The stirring was maintained, until it reached RT. Finally, the volume of the mixture was levelling to the initial volume by adding DI water. The resultant solution was denoted as PAT solution, and used for composition preparation.

2.6.1. Preparation of MXene–PAT polymer composite (MXPAT). The 25 mL of PAT solution and 1.2 g of MXene were stirred together for 24 h, and then the resultant solution was sprayed coated on FCCNW, which was initially dried using an air gun. Next, the dried composite was dehydrated in vacuum oven at 80°C for 5 h. After that, 50 mL of MXene colloidal solution was sprayed on the composite, and finally, the composite was dried for 1 h in vacuum oven. The final composite was utilized for additional analysis.

2.6.2. Preparation of MXene–PAT-conductive polymer composite (MXPAT-CP). The 10 mL of NMP, 0.795 g of conductive polymer, and 1.2 g of MXene were stirred together for 24 h. The mixture was transferred into 25 mL of PAT solution, and stirred for 1 day. The resultant solution was sprayed coated on FCCNW, which was dried using an air gun, and then dehydrated in vacuum oven at 80°C for 5 h. The 50 mL of MXene colloidal solution was sprayed on the composite, and dried for 1 h in vacuum oven. The corresponding composite of the polymers of PANI, PpAP, PANI-PpAP, pTh, and pPy were denoted as MXPATPN, PXPATPA, MXPATPNPA, MXPATPTh, and MXPATPTh, respectively. Then the composites were utilized for additional analysis.

2.6.3. Preparation of reduced MXene (r-MXene) and r-MXene–PAT composite (rMXPAT). First, 1 g of LiAlH4 and 5 g of MXene were mixed together in sufficient amount of tetrahydrofuran (THF) solution, and stirred for overnight. Then the resultant mixture was filtered, and washed with ample amount of THF. After that, the product was dried at 80°C in a vacuum oven for overnight. Next, the 1.2 g of r-MXene was mixed with 25 mL of PAT solution, and stirred for 24 h. The resultant dispersed solution was sprayed on FCCNW, and the composite was then dried at 80°C in a vacuum oven for 5 h. The surface coating was done by spraying 50 mL of MXene colloidal solution onto the composite, and finally dried for 1 h in vacuum oven. The composite was denoted as rMXPAT.

### Table 1: The composition of the PAT solution

| Compounds                  | Amount for 25 cm² |
|----------------------------|-------------------|
| Agar                      | 0.397 g           |
| PVA                       | 0.794 g           |
| Trimesic acid             | 39.675 mg         |
| Acrylic acid              | 0.198 g           |
| Glycerin                  | 1.984 mL          |
| PVDF in NMP (1 g L⁻¹)     | 1.984 mL          |
| Chitooligosaccharide       | 79.375 mg         |
| Polyacrylic acid          | 79.375 mg         |
| FeCl₃                     | 0.119 g           |
| DI water                  | 35.75 mL          |

2.7. Characterization

A field emission scanning electron microscope (SEM), S-4800 (Hitachi, Tokyo, Japan) was used to examine the surface morphology and cross-section of the composites. A High-power X-ray Diffractometer D/max-2500V/PC, (Ragaku, Tokyo, Japan) with Cu(Kz) was used to record the X-ray diffraction patterns of the composites. XPS with a (30–400) μm spot size at 100 W of Emax (Al anode) (K-Alpha, Thermo Fisher, East Grinstead, UK) was used to analyze the chemical environment and elemental percentage of the composites. The four-probe method FPP-RS8, DASOL ENG (Seoul, Korea) was used to measure the electric conductivity of the composites. The thermal stability of the composites was tested using a Thermal Analyzer DSC TMA Q400 (TA Instruments Ltd., New Castle, DE, USA). The thermal diffusivity, thermal conductivity, heat capacity and density were measured by using Laser Flash Apparatus LFA 467 (NETZSCH, Wittelsbacherstrabe, Germany). High-resolution Raman spectrophotometry (Jobin Yvon, LabRam HR Evolution (Hirorba, Tokyo, Japan)) was utilized to study structural features of the composites. A Mitutoyo thickness 2046S dial gage (Mitutoyo, Kanagawa, Japan) was used to measure the thickness of the composites. The X-band (8.2–12.4) GHz EM shielding was measured using a vector network analyzer (VNA, Agilent N5230A, Agilent Technologies, Santa Clara, CA, USA) with a sample size of 22.16 mm × 10.16 mm.

3 Results and discussion

3.1. Scanning electron microscope (SEM) analysis

The microstructure of Ti₃AlC₂ displayed metallic polycrystalline layered structure with higher aspect ratio. In general, MAX phase possesses hexagonal crystal structure with P6₃/mmc, where M–X forms an octahedral structure that is sandwiched between A layers. The cross section of the bulk Ti₃AlC₂ has smooth surface and layers are arranged like crystal rock (Fig. 3a). The morphology of MXene under SEM showed layers that were separated by gaps created by the etching process where A layers are eradicated. In addition, single layers of MXene were also observed in multilayer stack of MXene, which is due to the complete depletion of A layers. The cross section of MXene with a loosely stacked “accordion” structure indicates the successful occurrence of etching process and some individual flakes are observed on the surface of
MXene stack which confirms the complete separation of MXene layers from the stack (Fig. 3a, b and Scheme 1). The cross section of pure MXene film reveals that the edge curling is common and exfoliated MXene can join together and form MXene film which is mainly due to the hydrogen bond.\textsuperscript{14–27,41–43} The exfoliation process separates the layers of MXene stack, and forms single layers of MXene (Fig. 3c). The curling of the brink of single-layer MXene showed irregular shape, and the size also varied, even though a few small sized flakes which formed due to the curling, were spread over the large MXene flake. The exfoliated MXene undergone different heat treatment (evaporation and freezing) before filtration process (Scheme 1). According to the R. Bian et al. the freezing process induce the foxtail grass pattern in MXene composition which is agreed with Fig. 3c. According to our observation, the exfoliated MXene accumulates together, which cannot be used in solid composite preparation beforehand, and should be dispersed in a suitable solvent (Fig. 3c, and S1j of the ESI).\textsuperscript{44} The incessant exfoliation of MXene finally yields non-layered by-product, where we can predict the basic mechanism of MAX phase and MXene formation (Scheme 2). Further, the MXene made up of different size of layers and breaks in to the flakes during exfoliation process, and leaves the mark identified by Z on the origin. The marks X and Y indicate that cubic-like structure is formed during the MAX phase formation, and the exfoliation process exhibits the basement of the MAX phase. Further, corrosion also happens during the etching process (Fig. 3a and d). There are some cleavages present in the big by-product that is due to the etching process and settled down during centrifuging process, leaves exfoliated MXene flake in the supernatant (Fig. 3c and d). The supernatant is used to prepare d-MXene and these can be confirmed by using XRD in which the by-product resembled MAX phase, while possessing weak 002 peak at 9° confirmed presence of residual MXene (un exfoliated) (Fig. 8a).

The SEM images of conductive polymers shows different morphology (Fig. 4a–f). The PpAP image illustrates that the formation of highly porous interwoven network structure and resembles a cross-section of wood. The plate like irregular shape PANI, PANI–PpAP, and PPy stack together and seems like exfoliated d-MXene (Fig. 4b–d and S1j).\textsuperscript{36–38} The PTh possesses internal cavity within the particle, and small particles accumulate on the big particle, although obvious surface cavity is presented on the surface of the particle (Fig. 4e and f). In addition, the morphology of the conductive polymers depend the type of the oxidation agent and surfactant used.\textsuperscript{38} Moreover, the conductive polymers were invisible with the same morphology in the composites, in which it is believed that the conductive polymer mixed well with other ingredients in the composite.

The CCNW consists of irregularly arranged carbon fibre, which consists of cracks and defects. The cracks and defects are drawbacks, because they reduce the conductivity of carbon fabric. Hence, the EMI shielding of the composite is comparatively reduced. The functionalization of CCNW diminishes the defects, and enhances the conductivity and EMI shielding. In addition, the surface of the fibre of CCNW can be modified by various activities, such as coating of the carbon nanotube,
MXene, graphene, nanoparticle, and functionalization. In addition, the introduction of foreign nano-elements helps to interconnect the fibres, minimizes the defects, fills the cracks, and closes the cavity, which factors greatly influence the EMI shielding of the carbon fabric.\textsuperscript{45} Fig. 5 shows the topology of the composites, in which Fig. 5a, b, and f display coral-like structure, while Fig. 5c exhibits morphology like a bacterial colony. In general, the surface of the composite seems rough, due to the presence of MXene and polymers. The MXene are interconnected by polymers in which conductive polymer-based composite shows prominent differences in connection of MXene, compared with MXPAT. The MXPAT surface only consist the MXene flake while conductive polymers interconnect the surface MXene and form rough appearance. Nonetheless, the connection generates groves and cavities in the composite (Fig. S1a–i of the ESI\textsuperscript{†}). The polypyrrole-based MXene composite consists of hole which creates nest-like edifice and same structure is absent in other polymer composites. In addition, each hole has extra holes that indicate the structure of ant farm (Fig. 5e and S1e of the ESI\textsuperscript{†}). The d-MXene and PAT composition, rMXPAT and MXPATPN create surface cavities and grooves, whereas rMXPAT shows slightly deeper cavities, and MXPATPN have a narrow and deep groove structure, which contributes to a surface of multiple reflections (Fig. 5e and f, and S1e–i of the ESI\textsuperscript{†}). In general, the surface of dMXPAT seems like an amoeba, for which phenomenon d-MXene flakes are responsible for and formed surface non-continuous MXene film, create amoeba structure (Fig. S1h of the ESI\textsuperscript{†}).\textsuperscript{14,46,47} Furthermore, the MXene flake in the composite possesses similarity with exfoliated MXene, which indicates that the exfoliation happens during the composite preparation. The polymers, FeCl$_3$, DI water, and other components in the mixture induce the delamination process (Fig. 5a–f, and S1j–l of the ESI\textsuperscript{†}).
Fig. 6, and S2 of the ESI† show the cross-section of the composites. The CCNW is sandwiched between MXene-polymer composite (Fig. 6a, and S2a, d, and e†). The carbon fibres are arranged in parallel, and induce cavity formation in the circumference of fibres (Fig. S2a, d, e and g–i of the ESI†). The cross-section of the carbon fibre consists of delaminated MXene flakes that confirm the Raagulan et al. study, where MXene flake was intercalated with carbon nanotube, and possessed similar structural features (Fig. 6b and c). Furthermore, this implies the polymer composition based-exfoliation occurs during the MXene–PAT composition preparation. The cross-section of MXPATPy shows long tunnel, which indicates the internal structure resembles the ant farm, whereas rMXPAT composite consists of honeycomb cage (Fig. 6d and f). The composites exhibit many wave-like layers and F. Shahzad et al. reported that is due to the restacking of the MXene (Fig. 6e, and S2a–i of the ESI†), where multiple reflections are promoted (Section 3.6). The uMXPAT displays blade-like structure that is due to the presence of u-MXene, which was absent in the other composites (Fig. S2g)
of the ESIF). The cavities in uMXene leave continuous layered interfaces that end up in carbon fibre (Fig. 6e and S21 of the ESIF). Possessing the abovementioned structure and u-MXene leads to good thermal conductive material (Section 3.5).

3.2. X-ray photoelectron spectroscopy (XPS)

The XPS is used to explain the surface elemental composition, structures, defects, functional groups, and bonding nature of compounds. The Gaussian–Lorentzian function of Origin Pro® is utilized to plot the fitting and overlapping curve of MXene and composites. The fitting curve of MXene gives various information about the bonding nature and functional groups of MXene. The different peak positions of O 1s, F 1s, C 1s, and Ti 2p, of the Ti3C2Tx, are shown below (Fig. 7a-d). The functional groups, such as TiO2, C–Ti–Ox, Al2O3, and H2O of O 1s give rise to the peaks at (529.6, 531.1, 532.3, and 533.8) eV, respectively (Fig. 7a). The single peak of F 1s at 685.5 eV is due to the C–Ti–F functional group (Fig. 7b). The C–Ti–F, C–C, and CH–CO functional groups of C 1s engender the peaks at (281.1, 283.2, 284.5, and 286.1) eV which indicates the presence of the C–Ti and C–C bonds (Fig. 7c). The peak positions at (454.5, 456.4, 458.5, 461.3, and 464.5) eV for (Ti–C (2p3/2), Ti2+ (2p3/2), TiO2 (2p3/2), Ti2+ (2p1/2), and TiO2 (2p1/2)), respectively, are responsible for the functional groups attached to the Ti 2p environment, con- 

3.3. X-ray powder diffraction (XRD) analysis

The XRD profile is exploited to elucidate the crystalline or amorphous nature of the material based on peak position, height, and width. Fig. 8a and b show the XRD profiles of the Ti3AlC2, Ti3C2Tx, exfoliated MXene and composites, and draws a 2θ range between (5 and 90)°. The sharp peaks confirm that the MAX phase and MXene derivative are crystalline in nature and all the composite holding sharp peaks, implying the crystalline behavior of the composites. The crystalline MAX phase generates sharp 2θ peaks at 9.52° (002), 19.53° (004), 34° (101), 35.1° (102), 36.8° (103), 38.99° (008), 41.76° (104), 42.54° (105), 48.48° (107), 52.36° (108), 56.5° (109), 60.16° (110), 52.36° (1011), 64.98° (1011), 70.34° (1012), and 74.02° (118), and the interplanar crystal spacing of 2θ peaks at 9.52° is 9.72 Å. The etching process of the corresponding MAX phase eradicates or shifts peaks, and the sequence of new diffraction peaks is also formed. The formed MXene holds a crystalline nature, and the 2θ peak at 7.14° (002) is a characteristic peak of the MXene interplanar crystal spacing of 12.46 Å. In addition, the 2θ peaks initiating at 14.36° (111), 19.12° (200), 25.3° (101), 25.7° (110), 28.98° (220), 38.86° (222), and 40.9° (400) confirm the crystalline nature of the MXene, and attest to the occurrence of the etching process and formation of “accordion” structure. In addition, the etching process increase the interplanar space considerably. The peaks at 25.3° and 25.7° confirm the existence of TiO2 anatase and rutile, respectively (Fig. 3a and 5a). Moreover, peaks at 25.3° and 25.7° agree with XPS result (Fig. 7a-d). The exfoliation process mainly shifts the 7.14° peaks of MXene to lower refraction angles of (6.42° and 6.75°), which corresponds to d/f-MXene and d-MXene, respectively. Further, most of the peaks of both d/f-MXene and d-MXene overlap each other. In addition, d/f-MXene engenders non-overlapping weak peaks at (12.98, 19.69, 12.44, 22.88, 26.40, and 33.11)°, which differ from d/f-MXene and d-MXene. Therefore, the d/f-MXene and d-MXene can be differentiated by using these characteristic peaks. The XRD profile of u-MXene and MAX phase comprise almost similar Bragg’s angle, and the small humps at 2θ = 9.5° indicate that it holds slight MXene property. The SEM image of u-MXene confirms the above statement (Fig. 3d). Even though u-MXene does not display a layered structure (Fig. 3d), it cannot be used to synthesize MXene. The functionalization of CCNWT alters the peak position and erases some peaks of CCNW, where it shows shifts to higher binding energy of point P (Fig. 7e). These peaks are helpful to identify the temperature based process of MXene. The by-product (u-MXene) also exhibits similar behaviour of d-MXene under XPS, which can be further explained by using XRD (Section 3.3). The overlapping curve of F 1s, O 1s, and C 1s indicates different peak positions, which implies that the effective interaction occurred between polymers and MXene (Fig. 7g and h, and S3 and Tables S1 and S2 of the ESIF).
amorphous 2θ peak at 25.52°. In addition, the respective peak confirms the presence of carbon nanotubes and graphite structure in CCNW (Fig. S5c of the ESI†).

Fig. 8b displays the XRD graph of the composites, in which broad peaks and sharp peaks are observed. The broad peak at 20° is due to the FCCNW and agar in the composites, which
availability of PPy in MXPATPPy, and the combination of these distinctive Bragg angles at (18.4, 21.6, and 25.9)° indicate the presence of u-MXene, and other peaks confirm the u-MXene intercalates well with polymer and FCCNW matrix. The MXPAT, MXPATPN, MXPATPA, MXPATPNPA, MXPATPPPy, MXPATPTh, and uMXPAT own specific Bragg angles that are (9.79, 10.57, 9.88, 9.88, 9.94, 9.79, and 8.98)°, respectively. These slight differences guarantee that the interaction between polymer matrix and fillers was effective and can be identified the particular composite (Fig. 8b, S6a–h and Table S3 of the ESI†).

### 3.4. Raman spectroscopic analysis of composites

The Raman spectrum is the non-destructive and time-resolving tool that is utilized to investigate the structural feature, level of defects, crystalline nature, and physicochemical behaviour of the materials, such as graphene, MXene, carbon fabric, CNT, nanoparticles, and composites that can be studied using Raman spectroscopy. The characteristic peaks at (624 and 263) cm⁻¹ imply amorphous property remains in the composites, is due to the FCCNW [S5c and S6a–h of the ESI†], even though all the composites display many sharp peaks indicating the crystalline behaviour of the composites, which is due to the conductive polymers, MXene, and other composites. On the whole, the composites are a crystalline in nature and the peak at about 40° confirm the presence of MXene in all composites. The Bragg angles at (18.4, 21.6, and 25.9)° designate that PANI retains its identity in MXPATPN, and the weak 25.9° peak confirms the presence of orthorhombic unit cell with (200) reflective plane of PANI. The peaks at (19.6, 21.6, 23.9, and 25.8)° specify the existence of PpAP, in which the characteristic weak peak at 25.8° verifies the occurrence of stacks of poly aminophenol rings, and the hydrogen bond between MXene and polymers diminishes the intensity of the peak. The distinctive Bragg angles 2θ at (19.37, 20.47, 21.57, 22.67, and 25.3)° peaks confirm the PANI-PpAP copolymer. The typical diffraction angles at (19.53, 21.92, 25.3, and 25.9)° confirm the availability of PPy in MXPATPPPy, and the combination of these peaks suggest the amorphous nature of PPy. The broad peak that appears between (16.5 and 30)° assures the existence of polystyrene. The characteristic 2θ at (5.94, 8.98, 18.57, 19.84, and 38.38)° indicate the presence of u-MXene, and other peaks confirm the u-MXene intercalates well with polymer and FCCNW matrix. The MXPAT, MXPATPN, MXPATPA, MXPATPNPA, MXPATPPPy, MXPATPTh, and uMXPAT own specific Bragg angles that are (9.79, 10.57, 9.88, 9.88, 9.94, 9.79, and 8.98)°, respectively. These slight differences guarantee that the interaction between polymer matrix and fillers was effective and can be identified the particular composite (Fig. 8b, S6a–h and Table S3 of the ESI†).

![Fig. 8 XRD profile of (a) MXene, exfoliated MXene and MAX phase, and (b) composites.](image-url)
conductivity (Fig. 9b, S7c and d of the ESI†). The band in the 1602.49 cm\(^{-1}\) region engravens the stretching frequency of the C–C bond of PANI.\(^7\) In addition, the peaks in the range (800–1000) cm\(^{-1}\), such as (824.03 and 883.7) cm\(^{-1}\), affirm the presence of agar.\(^7\) The band at 1135.95 cm\(^{-1}\) can represent the stretching mode of bonds such as C–O and C–C of PVA. The peaks at (1179.66 and 1431.69) cm\(^{-1}\) represent the stretching frequency of –C–H and –OH of PVA (Fig. 9b, and S7a–h of the ESI).\(^7\) According to Kumar \textit{et al.}, the trimesic acid can be confirmed by characteristic peaks at 1648 cm\(^{-1}\)(C=O), 955 cm\(^{-1}\)(–OH–vibration) and 3085 cm\(^{-1}\)(–OH-stretching). The peak of C=O at 1651.9 cm\(^{-1}\) confirms the existence of trimesic acid, and the absence of the other two peaks shows that the trimesic acid bonds successfully with PVA and agar (Fig. 9b and S7c–e).\(^7\) In addition, the weak peaks of CCNW and MXene virtually disappear, which specifies that effective interaction has occurred among MXene, PANI, agar, PVA, and the other compositions (Fig. 9b, S7b–h and Table S4 of the ESI†). The MXPATPNPA overlaps with MXPATPN in about the 60 cm\(^{-1}\) range, while MXPATPA aligns in parallel up to 1000 cm\(^{-1}\), which designates the presence of PANI–PPyAP (Fig. 9b, and S7c of the ESI†). MXPATPN shows characteristic peaks within 1000 cm\(^{-1}\), and the MXPATPA also exhibits similar peaks at (419.35, 571.25, and 798.92) cm\(^{-1}\), even though MXPATPNPA do not reveal typical peaks in this region, but disclose distinctive peaks at 1341.16 cm\(^{-1}\). Further, the distinguishing peaks at 1603.93 cm\(^{-1}\) confirm the presence of MXPATPN, MXPATPA, and MXPATPNPA (Fig. 9b, and S7b–d†). The MXPATPNPy engenders distinctive bands at (924.8, 984.11, 1049.67, and 1566.89) cm\(^{-1}\), in which (924.8, 984.11, and 1049.67) cm\(^{-1}\) are considered as the ring deformation of quinoid polaronic, bipolaronic structure, and C–H in-plane deformation of pyrrole, respectively. The peaks at (701.24 and 1450.06) cm\(^{-1}\) confirm the C–C and C=C stretching frequency of MXPATPNPy.\(^7\) The range of (250–1500) cm\(^{-1}\) of MPAT, rMPAT, dMPAT, and uMXPAT exposes a linear trend, and exhibits distinctive peaks at (2225.06, 2228.48, 1761.03, and 2114.83) cm\(^{-1}\), by which the corresponding composites can be identified (Fig. 9b and S7†). Furthermore, the disappearance of the D and G bands of MXene and CCNW indicates that operative bonding and interaction have occurred between them, and formed perfect composite.

3.5. Thermal and electric property of the composites

3.5.1. Thermal stability and thermo gravimetric analysis of composites. The thermogravimetry analysis (TGA) and differential thermal analysis (DTA) were exploited to examine the thermal stability of the composite under a particular temperature range. The TGA and DTA analyses were carried out using Al\(_2\)O\(_3\) crucible in a nitrogen environment (flow rate of 50 mL min\(^{-1}\)) with a heating rate of 10 °C min\(^{-1}\). Fig. 11a and b show the TGA and DTG graphs, and all the composites display similar behaviour in both cases, with outstanding thermal stability. It was obvious that the weight loss of the composite varies, due to the constitutional composition effect of composites. The DTG plot displays several peaks that correlate with the endothermic and exothermic peaks of DSC, where the weight loss of the composite happens.\(^7\) The composite undergoes about 15% weight loss in the temperature range (30–175) °C. Even though the level of weight loss varies according to the constitutional components of the composites, MXPAT, MXPATPN, MXPATPNPA, MXPATPNPy, MXPATPNPTh and uMXPAT lose (16.81, 14.71, 13.37, 13.86, 14.33, 13.65, and 13.69) % of weight, respectively, and the corresponding temperatures are (194.17, 194.81, 190.77, 194.17, 180.14, 176.52, and 185.45) °C, respectively. The total reductions of weight of MXPAT, MXPATPN, MXPATPNPA, MXPATPNPy, MXPATPNPTh, and uMXPAT are (65.69, 58.86, 56.67, 55.09, 66.11, 63.54, and 59.05) %, respectively, in the entire temperature range of (30–1000) °C (Fig. 11a and b, and S8a–h and Table S5 of the ESI†). Among all the composites, MXPATPNPy shows a maximum of 66.11% of weight loss, while MXPATPNPA shows a minimum of 55.09%. The PANI–PPy co-polymer minimizes the thermal degradation, which is due to the hydroxyl and amine groups in polymer. On the other hand, the PPy also shows that nitrogen displays the highest degradation. Thus, these prominent differences are due to the copolymer effect. In addition, the thermal stability of the composite can also be explained by using heat-resistance
index \((T_s)\) which can be calculated from the TGA value of the composites. The \(T_{d5}\) (5% of weight loss) and \(T_{d30}\) (30% weight loss) can be derived from TGA and the \(T_s\) can be calculated by using the equation that is \(T_s = 0.49[T_{d5} + 0.6(T_{d30} - T_{d5})]\). The \(T_s\) reveals that how far the composite resists the thermal degradation and fire-retardant material have higher \(T_s\) value.\(^{26}\) The all the composites show the \(T_s\) value above \(~95^\circ C\) in which MXPATPA exhibits higher \(T_s\) value whereas MXPATPPy displays lower value. The \(T_s\) result shows that the MXPATPA resist thermal degradation more while MXPATPNPA exhibits less thermal degradation in entire temperature range (Fig. 10a and Table 2).

### 3.5.2. Thermal conductivity, thermal diffusivity, and heat capacity analysis of the composites.

There are many types of thermal conductivity measurement methods which can be categorized into two, those are steady-state measurement method and non-steady-state measurement method. The guarded hot plate method and guarded heat flow meter method are included in steady-state measurement methods while laser flash, transient plane source, temperature wave analysis, hot wire and line source probe method fall into non-steady-state method. In addition, the transient line source method is very fast method and can be used to measure molten and solid sample which is the advantage of transient line source method compare to another non-steady-state method stated. The steady state measurement is often time consuming and need demanding experimental conditions such as temperature control, specific dimension for sample and adiabatic environment. But the non-steady state measurement is more advantageous than steady-state measurement which is due to short measurement time, no rigorous environmental conditions and high accuracy.\(^{78}\) The thermal diffusivity, thermal conductivity, and heat capacity were obtained by NETZSCH LFA analyzer. For the investigation of thermal diffusivity, thermal conductivity, and heat capacity, LFA 467 instrument was used with 8.89 mm of spot size, diameter of 12.500 mm, LFA 467 steel furnace with 12.7 mm sample holder. All the analyses were performed under nitrogen environment with 60 mL min\(^{-1}\) purging rate and 20 mL min\(^{-1}\) protective flow rate. The thermal conductivity (TC) can be calculated according to eqn (1),\(^{79}\) where \(k\) is the thermal conductivity, \(m\) is the mass of the sample, \(C_p\) is the specific heat capacity of the sample, \((dT/dt)\) is the cooling rate, \(X\) is the thickness of the sample, \(A\) is the cross-section of the sample, and \(t_2\) – \(t_1\) is the temperature difference between the opposite surfaces of the sample. The SI unit of \(K\) is W (m K)\(^{-1}\). The thermal conductivity confirms through the various strategies that they are photon, phonon, and charge carriers. The high temperature gives rise to the photon conductivity, whereas atomic collision leads to a lattice vibration bringing about

![Fig. 10 TGA and differential thermal analysis (DTA) analysis of the composites (a) TG of composite, and (b) DTG of the composites.](image)

| Composites   | \(T_{d30}\) | \(T_{d5}\) | \(T_s\) (°C) |
|--------------|-------------|------------|--------------|
| MXPAT        | 252.78      | 127.59     | 99.33        |
| MXPATPN      | 254.30      | 128.26     | 99.90        |
| MXPATPA      | 259.58      | 131.71     | 102.13       |
| MXPATPNPA    | 257.64      | 130.71     | 101.37       |
| MXPATPPy     | 240.00      | 124.33     | 94.93        |
| MXPATPTh     | 247.92      | 122.22     | 96.84        |
| uMXPAT       | 262.22      | 119.44     | 100.50       |

![Fig. 11 Thermal diffusivity against diffusivity.](image)
phonon conductivity. The above-mentioned phenomena cannot act concurrently, even though one governs the others, which is determined by the atmosphere of the material used. In addition, TC can be indirectly estimated by using Fourier’s law, where $k$ is TC, $\alpha$ is the thermal diffusivity, $C_p$ is the heat capacity, and $\rho$ is the density of the material (eqn (2)). Further, the Parker formula is used to calculate the thermal diffusivity, where $d$ is the thickness of the sample, and $t_{0.5}$ is the time duration of the laser pulse to reach half of the maximum temperature (eqn (4)-(6)).

$$k = \frac{mc}{A(t_2 - t_1)}$$

$$k = \alpha C_p \rho$$

$$\alpha = 1.38 \frac{d^2}{t_{0.5}}$$

The thermal effusivity $e$ is the capability of transmission of the thermal energy to the surroundings, and is equivalent to the square root of the multiplication of the thermal conductivity, density, and heat capacity of the composite (eqn (4)-(6)).

$$C_p = \frac{\Delta H_m}{\Delta T}$$

$$e = \sqrt{k \rho C_p}$$

Table 1 shows the thermal diffusivity, thermal conductivity, heat capacity, thermal effusivity, density, and thickness of the composites, and the corresponding ranges are (0.185–0.282) mm$^2$ s$^{-1}$, (0.384–0.687) W (m K)$^{-1}$, (1.889–2.247) J (g K)$^{-1}$, (0.892–1.330) W s$^{1/2}$ (m$^2$ K)$^{-1}$, (0.988–1.336) g cm$^{-3}$, and (0.04–0.16) cm, respectively. The MXPATPPy and uMPAT display the minimum and maximum thermal diffusivity, whereas the corresponding thermal effusivity is (0.892 and 1.294) W s$^{1/2}$ (m$^2$ K)$^{-1}$, respectively. The 1.330 W s$^{1/2}$ (m$^2$ K)$^{-1}$ of MXPAT discloses the maximum thermal diffusivity, while 0.892 W s$^{1/2}$ (m$^2$ K)$^{-1}$ is shown by MXPATPPy. uMPAT reveals 78.9% of higher thermal conductivity, compared to MXPATPPy, even though MXPATPPy demonstrates 11.12% higher heat capacity, compared with uMPAT. The MXPATPPy shows lower density, which is directly correlated to the thermal diffusivity, thermal effusivity, and thermal conductivity, which show lower values of $\alpha$, $k$ and $e$ (Table 3). The $\alpha$ containing materials can be utilized for various applications, such as intelligent textiles, waste thermal harvest, thermal sensors, etc. The operative thermal energy garnering material should be phase-changing material (PCM), which can be confirmed by using thermal analysis, where exothermic and endothermic peaks can be predicted from the DTG profile. All the fabricated composites demonstrate phase changing ability and $e$ (Fig. 10b). Hence, the composites can be utilized for thermal energy storage purpose. The Fig. 10 shows that the uMPAT reveals higher $k$ and relatively higher $e$. uMPAT can be used for thermal storage, compared to the others. In addition, the by-product of MXene can be utilized for thermal energy storage process. In addition, the thermal energy can be converted into the electrical energy for which Thermoelectric Module (TEG) is used. The developed temperature variation creates the electrical potential between hot and cold junction which can be converted into electrical energy known as Peltier effect. This phenomenon can be explained by using seebeck concept that is $V = \alpha(T_h - T_c)$ where $V$ is voltage ($T_h - T_c$) temperature differences between hot and cold junction and $\alpha$ is seebeck constant. Further, the temperature variation between surfaces lead to charge carrier flow, produce the electricity. Hence, the thermal energy of the composite can be converted into electrical energy. The Kang et al. reported that $\alpha$ Ti$_3$C$_2$/epoxy (0.2 wt%) composite produced 0.122 mm$^2$ s$^{-1}$ of thermal diffusivity, and 0.270 W (m K)$^{-1}$ of thermal conductivity whereas pure epoxy displayed 0.110 mm$^2$ s$^{-1}$ and 0.243 W (m K)$^{-1}$ of thermal diffusivity and TC, respectively. The introduction of the MXene into the polymer enhance the thermal diffusivity and TC. Hence, the composites shown in Table 3 exhibit excellent thermal diffusivity and TC, compared to Ti$_3$C$_2$/epoxy composite. The in-plane TC of Ti$_3$C$_2$T$_x$ film discloses the (1.26 and 2.84) W (m K)$^{-1}$ at (80 and 290) K, respectively. The MXene/PVDF (5 wt%) manifests 0.363 of TC, which is lower than that of the composite fabricated in this study. Therefore, composite exhibit excellent thermal property and can be used as the thermal conductor (Table 3).

### 3.5.3. The electric conductivity (EC) of the composites.

The electric conductivity ($\sigma$) can be calculated from the factors such as resistance ($R$), resistivity ($\rho$), the cross-section ($A = \pi r^2$), where

| Composite | Diffusivity mm$^2$ s$^{-1}$ | Thermal conductivity W (m K)$^{-1}$ | Heat capacity J (g K)$^{-1}$ | Density g cm$^{-3}$ | Thermal effusivity W s$^{1/2}$ m$^2$ K$^{-1}$ |
|-----------|---------------------------|----------------------------------|-----------------------------|-------------------|----------------------------------|
| MXPAT     | 0.204                     | 0.600                           | 2.247                        | 1.312             | 1.330                            |
| MXPATPN   | 0.189                     | 0.524                           | 2.078                        | 1.336             | 1.206                            |
| MXPATPA   | 0.204                     | 0.504                           | 2.033                        | 1.217             | 1.177                            |
| MXPATPNPA | 0.229                     | 0.586                           | 1.997                        | 1.281             | 1.224                            |
| MXPATPPy  | 0.185                     | 0.384                           | 2.099                        | 0.988             | 0.892                            |
| MXPATPTh  | 0.198                     | 0.504                           | 2.028                        | 1.252             | 1.131                            |
| uMXPAT    | 0.282                     | 0.687                           | 1.889                        | 1.291             | 1.294                            |
$w$ is width and $t$, is thickness), length ($L$) and sheet resistance ($R_s$). The arithmetic multiplication of resistivity and length divided by cross-section gives rise resistance ($\rho = (\rho L)/A$) of the materials. The multiplied value of $R_s$ and $t$ produce the $\rho$. The conductivity of the material is reversely proposal to $\rho$ of the material.\(^4\) The thickness, electric conductivity, surface resistance and resistivity of the composites are shown in Table 4. The range of the thickness of the composites is 0.4–1.6 mm which indicates the thickness of the composite in $\mu$m range. The composites display 1.6–160 000 $\Omega$ and 0.128–256 000 $\Omega$ cm range of surface resistance and resistivity, respectively. The conductivity range of the composites place between 3.906 $\times$ $10^{-5}$ and 7.813 $S$ $cm^{-1}$. It was obvious that MXPATPPPy showed very higher resistance and resistivity compare to other composites and exhibits very low EC value of 3.906 $\times$ $10^{-5}$ $S$ $cm^{-1}$. Hence, the EMI shielding of the MXPATPPPy must be very low, though, it shows 99.99% of shielding efficiency which is discussed in details in Section 3.6. In addition, the MXPATPPPy exhibited low thermal diffusivity (0.185 mm\(^2\) $s^{-1}$), thermal conductivity (0.384 W (m K)$^{-1}$), density (0.988 g cm$^{-3}$) and thermal effusivity (0.892 W$^{1/2}$ m$^{-2}$ K$^{-1}$) which indicates that the lower EC and its structure influence the thermal properties of the MXPATPPPy (Tables 3 and 4, Fig. 5d, S1d and S2f). The internal channels in the composite and surface cavities play a major role for the uncommon behavior of the composite. EC of PVA is 1.63 $\times$ $10^{-12}$ $S$ $cm^{-1}$ and the EC of agar is 6.5 $\times$ $10^{-5}$ $S$ $cm^{-1}$ which imply that the addition of CP and MXene greatly improve the EC of the composites.\(^88\)\(^89\) Moreover, the EC of PANI,\(^8\) PTh,\(^4\) and PPy\(^4\) are 4.4, 7.17 $\times$ $10^{-6}$ and 0.61 $S$ $cm^{-1}$, respectively, and change in EC of the composite due to the presence of the MXene. The conductivity of MXPATPN and MXPATPA was lower than that of MXPAT, even though, MXPATPNPA exhibited 1000 times higher conductivity than MXPATPN which was due to the introduction of PNPA copolymer. Thus, PANI–PpAP copolymer have excellent conductivity and EMI shielding enhancing ability than PANI and PpAP (Table 4 and Section 3.6). The above mention phenomena happened due to the effective hole–electron transportation between MXene and PANI–PpAP. It is obvious that CP comparatively increase the EMI shielding of the composites (Section 3.6). The r-MXene and u-MXene composition held lower EC compare to MXene which is considered due to lack of proper orientation and bonding among constitutional elements, though, rMXPAT possessed 8.18 times higher conductivity than u-MXene. This is due to the availability of surface functional groups which promote bonding of fillers. The reduction process considerably replaced –F by –H in MXene and u-MXene significantly lost surface functional groups due to the exfoliation process. This process diminishes the charge flow along the matrix of the composite, thus, show lower (Table 4).

### 3.6. EMI shielding measurement

Electromagnetic radiation (EMR) is a transverse wave that can propagate through various materials. When EMR hits shielding materials, and undergoes transformation, such as absorption, reflection, and multiple reflections, it is called EMI shielding.

\[
\text{Transmittance} = \frac{P_{\text{out}}}{P_{\text{in}}} \\
\text{Reflectance} = \frac{P_{\text{refl}}}{P_{\text{in}}} \\
R + T + A = 1
\]

where, $R$, $T$, and $A$ imply the reflection, transmission, and absorption fractions, respectively. In addition, the absorption power can be articulated as $P_A = P_{\text{in}} - P_{\text{out}} - P_{\text{refl}}$. The total EMI shielding ($\text{SE}_T$) can be expressed by the logarithmic ratio between the power of incoming ($P_{\text{in}}$) and outgoing ($P_{\text{out}}$) EMR radiation (eqn (7)).

\[
\text{SE}_T = 10 \log\left(\frac{P_{\text{in}}}{P_{\text{out}}}\right) = \text{SE}_A + \text{SE}_R + \text{SE}_{\text{MR}}
\]

### Table 4 The electric conductivity comparison of the composites

| Composites     | Thickness (cm) | Surface resistance ($R_s$, $\Omega$) | Resistivity ($\Omega$ cm) | Conductivity (S cm$^{-1}$) |
|---------------|---------------|-------------------------------------|--------------------------|---------------------------|
| MXPAT         | 0.052         | 6.1                                 | 0.317                    | 3.153                     |
| MXPATPN       | 0.087         | 15                                  | 1.305                    | 0.00766                   |
| MXPATPA       | 0.062         | 13                                  | 0.806                    | 1.241                     |
| MXPATPNPA     | 0.08          | 1.6                                 | 0.128                    | 7.813                     |
| MXPATPPPy     | 0.16          | 160 000                             | 256 000                  | 3.906 $\times$ $10^{-5}$  |
| MXPATPTh      | 0.092         | 680                                 | 62.56                    | 0.0159                    |
| uMXPAT        | 0.04          | 270                                 | 10.8                     | 0.0926                    |
| dMXPAT        | 0.06          | 46                                  | 2.76                     | 0.362                     |
| rMXPAT        | 0.066         | 20                                  | 1.32                     | 0.758                     |
where, $SE_T$ is the total EMI shielding, $SE_A$ is the absorption, $SE_R$ is the reflection, and $SE_{MR}$ is multiple reflections. The effect of multiple reflections is insignificant, and the total EMI shielding can be expressed as $SE_T = SE_A + SE_R$. Further, the power of the incident wave ($P_i$) can be calculated by adding the reflected power ($P_a$), absorbed power ($P_a$), and transmitted power ($P_t$) that is $P_i = P_a + P_t + P_r$. In addition, the intensity can be calculated in the same way as $I_o = I_A + I_R + I_T$. The specific shielding effectiveness ($SSE$) is driven from the ratio between EMI SE and density ($\rho$) of the materials, and the corresponding unit is dB g$^{-1}$ cm$^{-3}$. It is obvious that low-density materials possess higher SE value. Furthermore, SSE does not have the thickness ($t$) basis information, because of the large $t$ with lower $\rho$ exhibits higher SSE. Hence, absolute effectiveness ($SSE/t$) with the unit of dB g$^{-1}$ cm$^2$ is used to evaluate the thickness effect of the material (eqn (8) and (9)).

\[
SSE = \frac{EMI \ SE}{\rho} \tag{8}
\]

\[
SSE/t = \frac{SSE}{t} \tag{9}
\]

The composites were fabricated by the spray coating process for which special polymer composition was synthesized, called PAT solution. The MXene was directly dispersed in PAT solution, while the conductive polymer was dispersed in NMP solution. Then both are mixed together, and finally utilized for the spray coating process on CCNW with areal density of 30 g m$^{-2}$. In the PAT solution preparation process PVDF used as a plasticizer, the trimesic acid is utilized to connect both agar and PVA for the desired FeCl$_3$ used as coupling reagent. The glycerin and polyacrylic acid were exploited as a dispersant and gained the strength of composite, respectively. The PAT is hydrophilic in nature, as it consists of agar and PVA. This PAT solution was a key player to disseminate MXene evenly, and conductive polymers intercalate within MXene, which creates a conductive network. In addition to the conductive network, CP plays a major role in creating the different structural features within the composite. It is virtually true according to the SEM that reflection ($SE_R$) and absorption ($SE_A$) give rise to new phenomena for EMI shielding. In the EMI shielding, the absorption plays a major role, while reflection exposes a general trend (Fig. 11a–c and Table S7 of ESf†). The maximum, minimum, and average SE ranges of composite are (39.33–45.18), (36.04–41.54), and (37.25–42.40) dB, while the maximum, minimum, and average $SE_R$ ranges are (4.71–11.44), (2.27–9.47), and (3.24–10.13) dB, respectively. In addition, the $SE_A$ is prominent compared to $SE_R$, and its minimum, minimum, and average ranges are (29.99–37.57), (26.24–36.46), and (27.55–36.77) dB, respectively. The maximum EMI shielding values of MXPAT, MXPATPN, MXPATPA, MXPATPNPA, MXPATPTh, uMXPAT, dMXPAT, and rMXPAT are (41.31, 39.33, 45.18, 44.08, 41.31, 42.99, 39.69, 43.74, and 44.31) dB, respectively, and the corresponding minimum EMI shielding values are (38.55, 37.83, 39.00, 37.87, 39.65, 40.25, 36.04, 41.54, and 39.94) dB, respectively. In these results, MXPATPA shows 45.18 dB of maximum shielding, whereas MXPATPN displays 39.33 dB of lowest EMI shielding which is due to the less conductive PANI. The minimum $SE_R$ values of MXPAT, MXPATPN, MXPATPA, MXPATPNPA, MXPATPTh, uMXPAT, dMXPAT, and rMXPAT are (9.27, 7.40, 8.41, 7.37, 2.27, 7.87, 9.18, 8.49, and 9.47) dB, respectively, while the corresponding maximum $SE_A$ values are (30.58, 30.44, 34.85, 36.56, 37.57, 34.87, 29.99, 34.48 and 34.09) dB, respectively. It was found that MXPATPA exhibits the higher SE of 45.18 dB, which is due to the PpAP, and the absence of PpAP reduces SE to about 10.94%. Hence, the intercalation of CP relatively alters the SE with respect to MXPAT. MXPATPNPA, MXPATPTh, and MXPATPN reveal (6.71, 0, 4.07, and –4.79)% respectively, where a minus sign indicates the reduction of SE with respect to MXPAT. The reduction of SE in MXPATPN due to PANI can be further affirmed by the Raman band at 1375 cm$^{-1}$ (Fig. 9b). It is apparent that the reduction of MXene increases SE by 7.26% which is another output that MXene can be reduced by LiAlH$_4$, and has an impact on EMI shielding (Fig. 11a, and Table S7 of ESf†). The MXPATPTh has ant farm-like structure, which promotes the absorption, and surface cavities can be seen by the naked eye. Fig. 11d below shows the basic mechanism of MXPATPTh, where surface cavities act as reflection, multiple reflections, and absorption units, while MXene layers contribute as multiple reflection units, which lead to absorption, and each anti-farm-like structure purely gives rise to absorption unit by internal multiple reflection process (Fig. 5d, S1e and S2e of ESf†). Hence, MXPATPTh demonstrates 23.14 times higher absorption compared to reflection. The MXPAT-PAPN shows 36.56 dB of absorption and 9.09 dB of reflection which is due to the porous conductive network of the MXPAT-PAPN. In addition, the MXPATPNPA displays higher EC of 7.813 S cm$^{-1}$ which is due to the presence of PpAP-PANI co-polymer, has impact on reflection process. In porous structure, PpAP-PANI co-polymer induces the reflection and it turn to multiple reflection in pores which leads to absorption. The basic mechanism is similar to the MXPATPTh (Fig. 12a–d and S2d, Table 4, S8 and S9 of ESf†). The dMXPAT and rMXPAT were studied to compare EMI shielding with MXPAT, in which rMXPAT shows higher EMI shielding of 44.31 dB, while MXPAT shows 41.31 dB with higher mechanical strength (2.88 MPa) and elastic nature compare to other composite fabricated (Fig. S9f†). Tensile strength of MXene is 1.8 MPa while MXene/eggshell membrane composite shows 1.48 MPa of tensile strength which is less than that of MXPAT (2.88 MPa) and MXene. In addition, the tensile strength of MXPATPTh and MXPATPN are 1.68 and 1.47 MPa, respectively, and tensile strength of MXPATPA is 4.3 MPa. The Nacre-Inspired Structure of MXene/cellulose composite displays 135.4 MPa of tensile strength. Hence, addition of PAT composition, arrangement of MXene flake in composite and presence of hydroxyl functional group in polymer significantly increase tensile strength of the composites. The SSE range is (28.85–40.49) dB g$^{-1}$ cm$^{-3}$, in which uMXPAT demonstrates the lowest value of 28.85 dB g$^{-1}$ cm$^{-3}$. The SSE/$t$ range was 236.45–721.25 dB g$^{-1}$ cm$^{-2}$ in which uMXPAT possessed maximum value of 721.25 dB g$^{-1}$ cm$^{-2}$ (Table S7f†).
The SE/t is calculated by normalizing the SE value by thickness which is density free comparison among SE values.\(^\text{95}\) The SE/t range of the composites is 1.393–1.6 dB mm\(^{-1}\) in which MXPATPA and MXPATPN possessed maximum and minimum value, respectively (Table S7†). All the composite showed 99.99% of EMI shielding effectiveness and corresponding bandwidth of MXPAT, MXPATPN, MXPATPA, MXPATPNPA, MXPATPy, MXPATPTTh, uMXPAT, dMXPAT and rMXPAT were (8.82–12.4, 8.2–12.4, 9.31–12.4, 8.2–11.98, 10.06–12.4, 8.42–12.36, 8.31–12.4, 8.24–9.72 and 8.24–11.19) GHz, respectively (Table S10†). Among all bandwidth uMXPAT held 4.09 GHz range bandwidth while dMXPAT possessed lowest of 1.48 GHz (Table S10†). The Shui. et al. study showed that Cu plate with a thickness of 3.1 mm exhibited 90 dB while Cu foil exhibited 70 dB with 0.01 mm of thickness.\(^\text{4,99}\) Despite the MXene/sodium alginate composition on PET film with the thickness of 45 μm gave rise 92 dB. Hence, MXene is effective protector than Cu in tiny electronic devises.\(^\text{28}\) The MXene/polyimide aerogel composite exhibits multifunctional properties and shows –45.4 dB of reflection loss in X-band region with 0.004 S cm\(^{-1}\) of EC, 2 mm of thickness and 0.009 g cm\(^{-3}\) of density while CNTs/ MXene/cellulose displays 38.4 dB of EMI SE with thickness of 0.004 cm and 25.066 S cm\(^{-1}\).\(^\text{55,99}\) This is obvious that the aerogel is an efficient structure for EMI SE of MXene polymer composite. The MXene coated cotton fabric shows 48.9 dB of EMI SE with 0.00067 cm of thickness, 2969 dB g\(^{-1}\) cm\(^{2}\) of SSE/t and 1570 S cm\(^{-1}\) of EC.\(^\text{49}\) In cotton fabric the MXene underwent wrinkling arrangement which increase the EMI shielding considerably. Thus, arrangement and order of MXene stacks are important for the EMI shielding behavior of the MXene composite.\(^\text{95,99,100}\) The carbon fabric with 30 g m\(^{-2}\) exhibited EMI shielding of 27.14 dB and Raagulan et al. study showed that the functionalization significantly changes the EMI shielding of carbon fabric.\(^\text{85,101}\) In addition, the comparison of previous work illustrated in Fig. 11f (Tables S11 and S12†). On the whole, the composites can be utilized both EMI shielding and thermal energy conversion process.

4 Conclusion

A new type of PAT composition was prepared at the optimized condition and which acts as a binder of MXene and conductive polymers. The FCCNW subjected as a template for spray coating process. The different strategy was utilized for mass production of exfoliated MXene and all form of MXene synthesized were utilized for composite preparation. The composites are light in weight, which possess thickness range of 0.04–0.16 cm and density laid between 0.988–1.336 g cm\(^{-1}\). In which MXPATPPy held lower density and uMXPAT exhibited lower thickness. The range of \(R_s\) is 1.6–160 000 Ω while MXPATPPy and MXPATPNPA showed higher and lower \(R_s\), respectively; corresponding \(\rho\) was 256 000 Ω cm and 0.128 Ω cm, respectively. On the other hand, MXPATPNPA showed an excellent EC of 7.813 S cm\(^{-1}\). Moreover, the composite had minimal thermal degradation till 175 °C. The thermal diffusivity range of the composites were 0.185–0.282 mm\(^2\) s\(^{-1}\) whereas composites hold 0.384–0.687 W (m K)\(^{-1}\), 1.889–2.247 J (g K)\(^{-1}\) and 0.892–1.330 W s\(^{1/2}\) m\(^{-2}\) K\(^{-1}\) range of thermal conductivity, heat capacity and thermal diffusivity, respectively. The uMXPAT had 0.282 mm\(^2\) s\(^{-1}\) and 0.687 W (m K)\(^{-1}\) of higher thermal diffusivity and thermal
conductivity, respectively. While MXPAT showed 2.247 J (g K)\(^{-1}\) and 1.330 W s\(^{1/2}\) m\(^{-2}\) k\(^{-1}\) of heat capacity and thermal effusivity, respectively. The MXPATPA exhibited maximum EMI shielding of 45.18 dB (9.31–12.4 GHz) with 33.26 dB g\(^{-1}\) cm\(^{3}\), 236.45 dB g\(^{-1}\) cm\(^{2}\) and 1.6 dB mm\(^{-1}\) of SSE, SSE/t and SE/t, respectively. The composites blocked almost 99.99% of incident radiation and maximum SE range of 45.18–39.33 dB while SSE, SSE/t and SE/t range were 27.99–40.49 dB g\(^{-1}\) cm\(^{3}\), 236.45–721.25 dB g\(^{-1}\) cm\(^{2}\) and 1.393–1.6 dB mm\(^{-1}\), respectively. The composites possess excellent EMI shielding ability, flexibility and good thermal behaviors. Thus, composites can be utilized for both thermal and EMI shielding applications.

Conflicts of interest
The authors declare no conflict of interest.

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