Structure and basic magnetic properties of the honeycomb lattice compounds

Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$

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Abstract

The synthesis, structure, and basic magnetic properties of Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ are reported. The crystal structures were determined by neutron powder diffraction. Na$_2$Co$_2$TeO$_6$ has a two-layer hexagonal structure (space group $P6_322$) while Na$_3$Co$_2$SbO$_6$ has a single-layer monoclinic structure (space group $C2/m$). The Co, Te, and Sb ions are in octahedral coordination, and the edge sharing octahedra form planes interleaved by sodium ions. Both compounds have full ordering of the Co$^{2+}$ and Te$^{6+}$/Sb$^{5+}$ ions in the $ab$ plane such that the Co$^{2+}$ ions form a honeycomb array; the stacking of the honeycomb arrays differ in the two compounds. Both Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ reveal antiferromagnetic ordering at low temperatures, with a metamagnetic transition displayed by Na$_3$Co$_2$SbO$_6$. 
Introduction

The study of the magnetic properties of low dimensional systems has played an important role in condensed matter physics.\(^1\) The understanding of both the theoretical models and the concepts of low dimensional magnets gained momentum with the discovery of high \(T_c\) superconductivity in cuprates, where the magnetic ions are arranged in square planes.\(^2\) Also of recent interest have been triangle-based geometries: when populated with antiferromagnetically coupled magnetic atoms, frustration of the low temperature magnetic states can occur due to conflicting near neighbor spin interactions.\(^3\) In this category, \(\text{Na}_x\text{CoO}_2\) represents one of the richest systems of current interest.\(^4-8\) The layered structures of the \(\text{Na}_x\text{CoO}_2\) phases consist of planes of edge-sharing cobalt-oxygen octahedra interleaved with Na ions.\(^9\) The Co ions have a triangular planar lattice, considered an important factor in driving the properties in these phases.

One variant of the triangular lattice is the honeycomb geometry, where 1/3 of the sites of a triangular lattice are non-magnetic. Compounds with the magnetic ions in this geometry have manifested interesting properties such as spin-glass behavior,\(^10\) spin-flop transitions\(^11\) and even superconductivity.\(^12\) Here we report the synthesis, structural and magnetic characterization of two Co-based compounds, \(\text{Na}_2\text{Co}_2\text{TeO}_6\) and \(\text{Na}_3\text{Co}_2\text{SbO}_6\), in which the Co ions are arranged in a honeycomb lattice. \(\text{Na}_3\text{Co}_2\text{SbO}_6\) was recently described as being trigonal\(^13\) but no other information related to the synthesis, structural or physical property characterization of this phase is available. The Te-based compound has not been previously reported.

Experimental
Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ were prepared by mixing Na$_2$CO$_3$ (Alfa, 99.5%) and Co$_3$O$_4$ (Alfa, 99.7%) with TeO$_2$ (JMC, 99.9995%) or Sb$_2$O$_4$ (Sb$_2$O$_5$ Alfa, 99.5%, annealed for 12 h at 1000°C in air to prepare Sb$_2$O$_4$). The Co:(Sb,Te) ratio was stoichiometric, but 20% excess Na$_2$CO$_3$ was added to compensate for loss due to volatilization. The mixtures were thoroughly ground, pressed into pellets ($\phi$=13 mm) and annealed under flowing nitrogen as follows: 8 days at 800°C were necessary for Na$_2$Co$_2$TeO$_6$ with two intermediate grindings - one after 4 days and then another one after 2 days of subsequent annealing; Na$_3$Co$_2$SbO$_6$ was obtained after 2 days of thermal treatment at 800°C. In all cases, the furnace temperature was increased by 5°C /min until the desired temperature was reached. Slow cooling to room temperature after reaction completion, 5°C /min, was performed. The compounds are both lightly pastel-colored, indicating that they have large band gaps and no significant electrical conductivity or Co mixed valency.

The purity of the samples was analyzed by powder X-ray diffraction using Cu K$_\alpha$ radiation and a diffracted beam monochromator. Neutron diffraction data were collected on Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ at the NIST Center for Neutron Research on the high resolution powder neutron diffractometer (BT1) with monochromatic neutrons of wavelength 1.5403 Å produced by a Cu(311) monochromator. Collimators with horizontal divergences of 15', 20' and 7' of arc were used before and after the monochromator and after the sample, respectively. Data were collected in the 2-theta range of 3º to 168º with a step size of 0.05º. The structural parameters were refined using the program GSAS.$^{14}$ The sodium content in each sample was determined by the structure refinement, and was in good agreement with that expected from nominal compositions.
The magnetic susceptibilities were measured with a Quantum Design PPMS system. Zero field cooled (ZFC) and field cooled (FC) magnetization data were taken between 1.8 K and 280 K in an applied field of 1 T. Field dependent magnetization data were recorded at 5 K and 30 K for Na$_2$Co$_2$TeO$_6$, and at 5 K, 8 K and 12 K for Na$_3$Co$_2$SbO$_6$.

**Results**

The purity of the compounds was confirmed by X-ray diffraction analysis. The powder neutron diffraction pattern of Na$_2$Co$_2$TeO$_6$ was indexed with a hexagonal cell in the space group $P6_322$ (No.182), while that of Na$_3$Co$_2$SbO$_6$ was indexed with a centered monoclinic cell in the space group $C2/m$ (No. 12). Structural analysis by the Rietveld method based on the powder neutron diffraction data was performed for both compounds. The cell constants, as determined by powder neutron diffraction, are presented in Tables 1 and 2. Na$_2$Co$_2$TeO$_6$ displays a primitive, two layer hexagonal structure, while Na$_3$Co$_2$SbO$_6$ has a single layer monoclinic structure, very similar to what is seen in Na$_3$Cu$_2$SbO$_6$.\(^{13}\)

The neutron powder diffraction pattern for Na$_2$Co$_2$TeO$_6$ shows it to have no observable dimensional deviations from hexagonal symmetry. Its honeycomb-based crystal structure was found to be very well described in space group $P6_322$. Two symmetry-independent Co ions within the honeycomb layer, one type of Te, one type of oxygen, and three independent Na ion positions were found. Refinements were carried out with atoms of the same type constrained to have the same thermal parameters. The Na distribution between the honeycomb layers is highly disordered. The geometry of the
oxygen array in the planes neighboring the Na planes creates triangular prismatic sites for the Na, exactly as is found in two-layer Na₃CoO₂⁹. In the Na₃CoO₂ structures, Na is found to occupy triangular prisms of two types: one type that shares only edges with the metal octahedra in the planes above and below, and one that shares faces with the metal octahedra in the planes above and below. The same situation is found for Na₂Co₂TeO₆. In this case, though, there are two types of face sharing possible, as the triangular prisms can share faces with one Co octahedron and one Te octahedron in the planes above and below, or two Co octahedra. The final refinements (Table 1) showed that the Na ions partially occupy all three types of available triangular prismatic sites, in different amounts: 71% of the Na are in the triangular prisms that share edges with octahedra in the honeycomb layers, 23% are in the prisms that share faces with the Co and Te, and 6% are in sites that share faces with two Co. The reason for this distribution is not understood, but as in the case of two-layer Na₃CoO₂, the fact that the face sharing positions are occupied at all indicates that the difference in energy that occurs on sharing faces with the octahedra in neighboring planes is not as important as other factors such as Na-Na interactions in determining the Na distribution. Again as in the case of Na₃CoO₂,¹⁵,¹⁶ some of the Na ions were found to be displaced towards the faces of their triangular prisms. Refinements in which the Na were left in the centers of their coordination polyhedra resulted in worse agreement factors (χ²=1.47, wR_p=6.72, R_p=5.26) than for the final model, and physically unrealistic thermal parameters for the Na. It was found that the displacement of Na(2) did not significantly improve the agreement factors or the thermal parameters, so in the final refinement, only Na(1) and Na(3) were allowed to relax from their ideal positions; they were found to displace towards the faces of the prisms. The freely refined occupancies of
the three Na sites, which are all partially occupied, yield a formula within experimental uncertainty of the ideal Na₂Co₂TeO₆ composition. The final model provides an excellent fit to the data. The observed, calculated, and difference plots for the refinement based on this model are presented in Figure 1. The refined structural parameters are presented in Table 1. The Na distribution is shown in Figure 4.

A classic two-layer structure, similar to that of Na₀.₇CoO₂,⁹ but with a larger in-plane unit cell to accommodate the honeycomb-type cobalt-tellurium ordering is found for Na₂Co₂TeO₆. No Co-Te intermixing was found for the metal atom sites: the atoms are fully ordered. The crystal structure is presented in Figure 3 a and b. Selected bond distances are shown in Table 3. The two independent Co sites have very similar Co-O bond distances within their octahedra, 2.11 Å and 2.13 Å for Co(1) and Co(2), respectively. The Te⁶⁺O₆ octahedron is significantly smaller (Te-O bond length = 1.95 Å), as expected. The bond valence sums are consistent with the assignment of Co²⁺ and Te⁶⁺ as formal oxidation states. The two-layer stacking sequence found for Na₂Co₂TeO₆ is distinctly different from that found for the Cu analog, Na₂Cu₂TeO₆,¹⁷ which, though monoclinic, is based on a pseudo-hexagonal three-layer stacking sequence.

Na₃Co₂SbO₆ was found here to be isostructural with Na₃Cu₂SbO₆. It has a single layer structure, well described in the C2/m space group. The structure displays one type of Co, one type of Sb, two independent sodium ions and two different oxygen sites. The refinement, using starting positions⁹ from Na₃Cu₂SbO₆ quickly converged to a final structural model. No mixing was found among the Co and Sb, and the freely refined Na site occupancies were found to be within experimental uncertainty of full occupancy, yielding the ideal composition. All site occupancies were therefore set to 1 in the final
refinements. The final structural parameters are presented in Table 2. The observed, calculated, and difference plots for the neutron diffraction data, using the final model, showing the excellent fit of the model to the data, are presented in Figure 2. The crystal structure is presented in Figure 3 a and c. Selected bond distances are shown in Table 3. The monoclinic symmetry allows for the possibility of distortion of the Co\(^{2+}\)-O octahedron, but the six Co-O distances are in the range of 2.12-2.13 Å, showing that no substantial distortion has occurred. The Co-O bond lengths are very similar to those observed for Na\(_2\)Co\(_2\)TeO\(_6\) (2.11-2.13 Å). The SbO\(_6\) octahedron in Na\(_3\)Co\(_2\)SbO\(_6\) has a size (Sb\(^{5+}\)-O distances 2.03-2.06 Å) very similar to that found for the Co-O octahedron. The bond valence sums are consistent with the assignment of Co\(^{2+}\) and Sb\(^{5+}\) as formal oxidation states.

The temperature- and field-dependent magnetization data for Na\(_2\)Co\(_2\)TeO\(_6\) and Na\(_3\)Co\(_2\)SbO\(_6\) are shown in Figures 5-8. Both compounds display Curie-Weiss behavior above approximately 150 K. Deviations from the high temperature behavior at lower temperatures suggest the presence of either low dimensional ordering or short range magnetic fluctuations (Figures 5 and 6). The magnetic moments inferred from the linear fits of the inverse susceptibility data are \(\mu_{\text{eff}} = 4.97 \, \mu_B\) and \(\mu_{\text{eff}} = 4.90 \, \mu_B\) per Co for Na\(_2\)Co\(_2\)TeO\(_6\) and Na\(_3\)Co\(_2\)SbO\(_6\) respectively. These moments are in agreement with the commonly observed value of 4.8 \(\mu_B\) for Co\(^{2+}\) ions in a high spin \(d^7\) configuration.\(^{18}\)

The temperature dependent magnetization data for Na\(_2\)Co\(_2\)TeO\(_6\) show a maximum near 20 K (top inset in Figure 5), characteristic of a magnetic transition that is predominantly antiferromagnetic (AFM) in character. From the maximum in \(d(M^*T)/dT\) (lower inset in Figure 5), the Néel temperature \(T_N\) is determined to be 18 K. A second
transition near 9 K is also apparent in the derivative data, which may be associated with spin reorientation. Both transition temperatures are marked by vertical arrows in the figure. Although the positive Weiss temperature $\theta_W$ ($\theta_W = 7$ K) suggests the presence of weak ferromagnetic (FM) interactions, the magnetization versus magnetic field (M vs. H) data recorded at 5 K and 30 K (Figure 6) give no clear evidence of a hysteresis loop. The curve at 5 K has an upward curvature, however, confirmed by its derivative (inset to Figure 6), suggesting the presence of a broad metamagnetic transition. The fact that this upward curvature is not present in the 30 K data indicates that at this temperature the system is in the paramagnetic state, consistent with the temperature-dependent data and the $T_N$ value of 18 K.

The temperature- and field-dependent magnetization data for Na$_3$Co$_2$SbO$_6$ are presented in Figures 6 and 8. The ZFC/FC susceptibility data in an applied magnetic field $H = 1$ T are indistinguishable and display a broad maximum below 5 K (top inset in Figure 6). This again is indicative of a primarily AFM transition, with the ordering temperature determined from the maximum in $d(M*T)/dT$ to be $T_N = 4.4$ K, despite a positive Weiss temperature $\theta_W = 15.6$ K. The M(H) data (Figure 8) show no hysteresis, but do clearly indicate the existence of a metamagnetic transition that broadens and moves to lower fields as T increases (inset to Figure 8). The critical field values for the transition are seen in the maxima in the $dM/dH$ plots shown in the inset. A field of only 1 T is required to make the Co spins reorient at $T = 5$ K.

**Discussion and Conclusions**
The crystal structures of Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ consist of planes of edge-sharing octahedra of CoO$_6$ mixed with TeO$_6$ and SbO$_6$ octahedra, respectively, in an ordered fashion: the Co$^{2+}$ ions form a honeycomb lattice, where each Co has three Co near neighbors, and the non-magnetic Te$^{6+}$ or Sb$^{5+}$ ions sit in the middle of each Co honeycomb. The projection along the $c$ axis of Na$_2$Co$_2$TeO$_6$ shown in Figure 3 illustrates the type of network formed by Co and Te/Sb in the $ab$ plane in both compounds. This ordered distribution of the metallic ions results in an enlargement of the in-plane hexagonal cell dimension of Na$_2$Co$_2$TeO$_6$ over that of the simple triangular array, $a_{\text{HEX}}$, to $a_{\text{HEX}} \times \sqrt{3}$. The $ab$ plane is dimensionally hexagonal in Na$_3$Co$_2$SbO$_6$ as well, though the overall symmetry of the structure is monoclinic.

In Na$_3$Cu$_2$SbO$_6$ and Na$_2$Cu$_2$TeO$_6$, due to the Jahn-Teller effect manifested by the Cu ions, the Cu honeycomb lattice is quite distorted, and can be considered a lattice of Cu dimers.$^{13,15}$ No such distortion is found in the Co honeycomb network for either Na$_2$Co$_2$TeO$_6$ or Na$_3$Co$_2$SbO$_6$, though in the latter case it would be allowed by symmetry. Slightly smaller honeycombs are found in Na$_2$Co$_2$TeO$_6$ than in Na$_3$Co$_2$SbO$_6$ due to the smaller size of the Te$^{6+}$ ion.

The sodium distribution found in Na$_2$Co$_2$TeO$_6$ is different from that in Na$_3$Co$_2$SbO$_6$. In Na$_2$Co$_2$TeO$_6$, a disordered Na lattice arrangement is seen, whereas in Na$_3$Co$_2$SbO$_6$ the Na ions are ordered in a simple, completely filled triangular array. Although the average structure shows the Na atoms to partially occupy sites between the layers in a disordered fashion in Na$_2$Co$_2$TeO$_6$, short range ordering may be present.

The most important difference between the two structures is the stacking of the honeycomb arrays. In the case of Na$_2$Co$_2$TeO$_6$, the Co honeycomb arrays are staggered
(Figure 9a) while in Na$_3$Co$_2$SbO$_6$ the Co honeycomb arrays stacked directly on top of each other (Figure 9b) similar to the Cu honeycomb planes in Na$_3$Cu$_2$SbO$_6$. The difference in stacking of the honeycomb layers is expected to strongly impact the magnetic interactions.

The magnetic susceptibility data for both compounds are consistent with the presence of Co$^{2+}$ ions in the high spin configuration (S=3/2). The in-plane Co-Co distances within the honeycomb for the two compounds are similar (3.052 Å for Na$_2$Co$_2$TeO$_6$ and 3.098 Å for Na$_3$Co$_2$SbO$_6$), and the superexchange pathways, through ~90 degree Co-O-Co bonds between edge-sharing octahedra, are also very similar. Thus, it is not surprising that the Curie Weiss thetas are ferromagnetic in nature and similar in magnitude (7K for Na$_2$Co$_2$TeO$_6$ and 15.5K for Na$_3$Co$_2$SbO$_6$). The compounds are not isomorphous, however, due to differences in the interplanar stacking. This may be the origin of the different magnetic ordering temperatures (18 K for the Te compound and 4.4 K for the Sb compound). The difference in stacking may also lead to the difference in behavior in an applied magnetic field. Clear evidence for a metamagnetic transition is seen in Na$_3$Co$_2$SbO$_6$ at low fields but the situation is less clear in the Te compound, where there is only a broad increase in the magnetization with applied field.

BaCo$_2$(AsO$_4$)$_2$ and BaCo$_2$(PO$_4$)$_2$ are examples of well studied systems in which the Co$^{2+}$ ions are also located on a honeycomb lattice. These systems represent classical examples of 2D XY magnetic models, and have different magnetic ground states. Some of the magnetic properties observed for the current compounds, i.e. the metamagnetism, are reminiscent of what is seen in those materials. A detailed study of the magnetic behavior of single crystals of Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ would be of
particular interest because their honeycomb lattices are similar in size and geometry but differ in stacking. Such a comparison might illuminate the effects of interlayer interactions on the low temperature magnetic state of honeycomb lattices.

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Table 1. Crystallographic data for Na$_2$Co$_2$TeO$_6$ in the space group $P\ 6_3\ 2\ 2$ (No. 182), $a=5.2889(1)$Å, $c=11.2149(4)$Å, Volume = 271.68(1) Å$^3$.

| Atom  | Wyckoff position | x          | y          | z       | $U_{iso}\times100$ | Occ |
|-------|------------------|------------|------------|---------|--------------------|-----|
| Co(1) | 2b               | 0          | 0          | $\frac{1}{4}$ | 0.8(1)            | 1   |
| Co(2) | 2d               | $\frac{2}{3}$ | $\frac{1}{3}$ | $\frac{1}{4}$ | 0.8(1)            | 1   |
| Te    | 2c               | $\frac{1}{3}$ | $\frac{2}{3}$ | $\frac{1}{4}$ | 0.9(1)            | 1   |
| O     | 12i              | 0.6432(6)  | -0.0252(4) | 0.3452(1) | 1.50(4)           | 1   |
| Na(1) | 12i              | 0.24(1)    | 0.65(2)    | -0.026(3) | 0.8(3)            | 0.081(7) |
| Na(2) | 2a               | 0          | 0          | 0        | 0.8(3)            | 0.11(3) |
| Na(3) | 12i              | 0.702(3)   | 0.066(2)   | -0.006(2) | 0.8(3)            | 0.234(6) |

$\chi^2=1.39; \ wRp=6.53; \ Rp=5.05$
Table 2. Crystallographic data for Na₃Co₂SbO₆ in the space group C 2/m (No. 12),
\(a=5.3681(2)\)Å, \(b=9.2849(4)\)Å, \(c=5.6537(2)\)Å, \(\beta=108.506(4)^\circ\), Volume= 267.22(2) Å³.

| Atom | Wyckoff position | x    | y    | z    | Uiso*100 | Occ |
|------|------------------|------|------|------|----------|-----|
| Co   | 4g               | 0    | 2/3  | 0    | 0.23(12) | 1   |
| Sb   | 2a               | 0    | 0    | 0    | 1.86(17) | 1   |
| O(1) | 8j               | 0.2744(8) | 0.3410(3) | 0.7971(5) | 1.25(2) | 1   |
| O(2) | 4i               | 0.2463(10) | 0.5 | 0.2056(10) | 1.25(2) | 1   |
| Na(1) | 2d            | 0    | 0.5  | 0.5  | 1.58(5)  | 1   |
| Na(2) | 4h            | 0.5  | 0.3296(9) | 0.5 | 1.58(5)  | 1   |

\(\chi^2=1.01;\ wR_p=5.31\%;\ Rp=4.31\%\).
Table 3. Selected bond distances for Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ at room temperature. BVS= bond valence sum.

|                  | Na$_2$Co$_2$TeO$_6$       | Na$_3$Co$_2$SbO$_6$ |
|------------------|---------------------------|---------------------|
| Co(1)-O×6        | 2.112(3)                  | Co-O(1)×2           | 2.132(4) |
| Co(2)-O×6        | 2.131(3)                  | Co-O(2)×2           | 2.124(3) |
| BVS              | 1.93 (Co(1))              | Co-O(1)×2           | 2.125(4) |
|                  | 1.83 (Co(2))              |                     | 1.85     |
| Te-O×6           | 1.951(2)                  | Sb-O(1)×4           | 2.026(3) |
| BVS              | 5.98                      | Sb-O(2)×2           | 2.057(6) |
|                  |                           |                     | 5.10     |
| Na(1)-O*         | 2.50(5)                   | Na(1)-O(1)×4        | 2.367(4) |
| Na(2)-O          | 2.515(3)                  | Na(1)-O(2)×2        | 2.427(6) |
| Na(3)-O*         | 2.49(1)                   | Na(2)-O(1)×2        | 2.368(4) |
|                  |                           | Na(2)-O(2)×2        | 2.431(6) |
|                  |                           |                     | 2.385(6) |
| Co-Co            | 3.054 ×3                  |                     | 3.095 ×1 |
|                  |                           |                     | 3.098 ×2 |

* The bond length is averaged over 6 different bonds.
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Figure Captions:

**Figure 1.** Observed (crosses) and calculated (solid line) neutron diffraction intensities for Na$_2$Co$_2$TeO$_6$ at 295K. Vertical bars show the Bragg peak positions. Regions in the pattern in the vicinity of two impurity peaks (39.6-40.7 and 59.8-60.1 degrees) were omitted from the refinement. The difference plot is shown at the bottom.

**Figure 2.** Observed (crosses) and calculated (solid line) neutron diffraction intensities for Na$_3$Co$_2$SbO$_6$ at 295K. Vertical bars show the Bragg peak positions. The difference plot is shown at the bottom.

**Figure 3.** The crystal structures of Na$_2$Co$_2$TeO$_6$ and Na$_3$Co$_2$SbO$_6$ (a) The layer of edge-shared MO$_6$ octahedra, showing the honeycomb Co array.  (b) View perpendicular to the honeycomb layers in Na$_2$Co$_2$TeO$_6$ (c) view perpendicular to the honeycomb layers in Na$_3$Co$_2$SbO$_6$. Inside the octahedra, small dark grey spheres represent Co while bigger black spheres represent Te. The sodium ions are represented by light gray spheres. Oxygens are at the vertices of the octahedra.

**Figure 4.** The Na distribution in Na$_2$TeCo$_2$O$_6$. All Na positions are partially occupied. The oxygens in the layers above and below are shown as red spheres. They form three types of triangular prismatic sites for Na. These are marked with their triangular bases in the figure: Na(1) shares faces with one Co and one Te octahedron, and contains approximately 23% of the Na; Na(2) shares faces with two Co octahedra, and contains approximately 6% of the Na; and Na(3) shares only edges with the Co and Te octahedra in the layers above and below (approximately 71% of the Na). Na(1) are grey, Na(2) cyan and Na(3) blue spheres, respectively.
Figure 5. The temperature dependence, at an applied field of $H = 1$ T, of the magnetization for $\text{Na}_2\text{Co}_2\text{TeO}_6$. The dashed line shows the fit of the high temperature data to a Curie-Weiss law. The insets are: upper – detail of low temperature molar susceptibility; lower – low temperature $d(MT)/dT$ at $H = 1$ T. Significant features at 18 and 9 K are seen in the derivative data.

Figure 6. The temperature dependence, at an applied field of $H = 1$ T, of the magnetization for $\text{Na}_2\text{Co}_2\text{SbO}_6$. The dashed line shows the fit of the high temperature data to a Curie-Weiss law. The insets are: upper – low temperature susceptibility at $H = 1$ T; lower – low temperature $d(MT)/dT$ at $H = 1$ T: a feature is seen at 4.4K.

Figure 7. The field dependence at $T = 5$ K and 30 K of the magnetization for $\text{Na}_2\text{Co}_2\text{TeO}_6$. The inset shows the magnetization derivative $dM/dH$ vs magnetic field at $T = 5$ K and 30 K.

Figure 8. The field dependence at $T = 5$ K, 8 K and 12 K, of the magnetization for $\text{Na}_2\text{Co}_2\text{SbO}_6$. The inset shows the magnetization derivative $dM/dH$ at $T = 5$ K, 8 K and 12 K. A metamagnetic transition is clearly seen.

Figure 9. View of adjacent Co-M (M=Te, Sb) planes in $\text{Na}_2\text{Co}_2\text{TeO}_6$ and $\text{Na}_3\text{Co}_2\text{SbO}_6$, showing the stacking of the cobalt honeycomb lattices. In both cases, the planes have been tilted slightly so that all the metal atoms are visible: for the Te compound the metals are directly on top of each other in the direction perpendicular to the plane while in the Sb case they are shifted slightly. For $\text{Na}_2\text{Co}_2\text{TeO}_6$ (a), the stacking of the honeycomb Co lattices in neighboring planes is staggered, while, for (b) $\text{Na}_3\text{Co}_2\text{SbO}_6$, the Co honeycombs stack nearly directly on top of one another. Small dark grey spheres represent Co, bigger black spheres are Sb and big dark gray crossed spheres are Te.
Figure 1
\begin{figure}
\centering
\includegraphics[width=\textwidth]{Na3Co2SbO6}
\caption{Na$_3$Co$_2$SbO$_6$}
\end{figure}
Figure 4
Figure 5

Na$_2$Co$_2$TeO$_6$

$H = 1\ T$

$M/H$ (emu/mole Co)

$T$ (K)

$H/M$ (mole Co/emu)

$T$ (K)

$d(MT)/dT$

$T$ (K)

9.0 K

17.7 K
Figure 6
Figure 7
Figure 8
(a) Na$_2$Co$_2$TeO$_6$

(b) Na$_3$Co$_2$SbO$_6$

Figure 9