A Review of Proximity Gettering Technology for CMOS Image Sensors Using Hydrocarbon Molecular Ion Implantation

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We developed a high-gettering-capability silicon wafer for advanced CMOS image sensors using hydrocarbon molecular ion implantation. We found that this novel silicon wafer has an extremely high gettering capability for metal, oxygen, and hydrogen impurities during the CMOS device fabrication process. We also found that the white spot defect density of a hydrocarbon-molecular-ion-implanted CMOS image sensor was substantially lower than that of a CMOS image sensor without hydrocarbon molecular ion implantation. This indicates that the novel silicon wafer helped improve device performance parameters such as white spot defect density and dark current. We believe that this wafer will be beneficial in the design of silicon wafers for advanced CMOS image sensor fabrication.

1. Introduction

CMOS image sensors have become widely used in Internet of Things (IoT) devices such as smart phones, smart watches, and personal computer tablets. The consumer market for such devices strongly requires a wide dynamic range, highly sensitive imaging, and high-speed image data processing for fabricating high-performance CMOS image sensors. However, there are serious technical issues involved in fabricating them.

The first important issue is metallic impurity contamination in the device active region during CMOS device fabrication process steps such as high-temperature rapid thermal treatment and plasma etching treatment. Metallic impurities form deep energy level defects in the silicon band gap. These defects strongly affect device electrical performance parameters such as white spot defect density, recombination lifetime, dark currents, and gate oxide breakdown voltage. Thus, CMOS image sensor manufacturers make much effort to eliminate metallic impurities from the device active region using gettering techniques. Intrinsic gettering (IG) is the most popular gettering technique in the semiconductor device fabrication process. This technique involves gettering metallic impurities by using an

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oxygen precipitation growth process during the device heat treatment. For advanced CMOS devices, thermal heat budgets are a much lower temperature and a shorter duration time year by year. Thus, oxygen precipitation is not enough for the growth of oxygen precipitate nucleation center, which can act as gettering sinks for metallic impurities during the device fabrication process. Thus, IG becomes unsuitable for fabricating advanced CMOS image sensors. Another gettering solution is high-energy ion implantation induced by damage-induced defects for gettering sinks in the silicon substrate. It is well known that these defects induce secondary extended defects such as dislocations and dislocation loops after the device heat process. The secondary extended defects can act as extremely effective gettering sinks for metallic impurities in the silicon substrate. However, these defects expand to the top surface region of the silicon substrate, which makes it extremely difficult to grow epitaxial silicon in the region. Thus, the high-energy ion implantation technique is not a good gettering technical solution for fabricating advanced CMOS image sensors.

The second important issue is oxygen out-diffusing to the device active region from a Czochralski (CZ)-grown silicon substrate during the CMOS device heat process. The CZ-grown silicon crystals produced by a crystal growth process using a quartz crucible contain high oxygen impurities in bulk silicon. The oxygen impurities form oxygen-related deep energy level defects in the silicon band gap, which potentially can act as potential pockets and barriers in the device active region. These defects strongly affect device electrical performance factors such as image lag characteristics. Thus, CMOS image sensor manufacturers strongly demand the control of oxygen impurities out-diffusing to the device active region from a CZ-grown silicon substrate during the device fabrication process. Thus, the solution to the above issue is the use of CZ-grown silicon wafers with low oxygen concentration. The out-diffusion amount of such wafers decreases during the device fabrication process. However, the mechanical strength of the silicon wafers is much lower than that of conventional CZ-grown silicon crystals. Thus, it is extremely difficult to use these wafers for CMOS image sensor mass production.

The third important issue is the carrier generation from interface state defects such as the presence of transistor gate oxide in the Si/SiO₂ interface region, local oxidation in silicon (LOCOS),
shallow trench isolation (STI), and deep trench isolation (DTI).\(^{(1,2)}\) These carrier generations strongly affect electrical device performance parameters such as the levels of fixed pattern noise, flicker noise, and random telegram signal (RTS) noise in CMOS image sensors.\(^{(20,21)}\) Therefore, CMOS image sensor manufacturers make strong effort to reduce the interface state density using low-temperature hydrogen annealing treatment. This treatment decreases the Si/SiO\(_2\) interface state density of Pb and E’ centers by using a hydrogen-terminated process.\(^{(22)}\) However, the three-dimensional stacked CMOS image sensor (3D-CIS) fabrication process has been using multilayer deposition on the device surface fabricated by using the atomic layer deposition technique.\(^{(22)}\) The device surface deposits on multiple dielectric layers are performed before low-temperature hydrogen annealing. Most of the hydrogen atoms are trapped in the multiple dielectric layers during the low-temperature hydrogen annealing.\(^{(23)}\) The hydrogen does not diffuse to the device active region in the CMOS image sensor. The hydrogen amount is insufficient for process-induced defect (interface state defect) passivation by hydrogen. Therefore, CMOS image sensor manufacturers have been demanding an alternative to low-temperature hydrogen annealing for effectively decreasing the process-induced defect (interface state defect) density in the device active region. The CMOS image sensor manufacturers strongly require silicon wafer solutions to address the above three technical issues for achieving a high performance in CMOS image sensor manufacturing.

How can we address these technical issues associated with CMOS image sensor manufacturing? The gettering technique is considered to be the most critical solution of the above issues with advanced CMOS image sensor manufacturing. The considered gettering technique forms gettering sinks near the device active region through ion implantation and can be called proximity gettering. The advance CMOS device manufacturing thermal budget tended to be lower than that in the conventional CMOS device manufacturing thermal budget. Thus, when slowly diffused metallic impurities such as Fe, W, and Mo do not diffuse to the gettering sinks in the epitaxial silicon wafer substrate such as IG technique during the low-temperature device heat process, it is extremely difficult to obtain a high gettering capability for slowly diffused metallic impurities using a conventional IG technique. Therefore, we have been developing an alternative silicon wafer gettering technology solution for advanced CMOS image sensor fabrication using hydrocarbon molecular ion implantation.\(^{(24)}\) This novel proximity gettering technology realized the formation of a gettering sink near the device active region. We call it a hydrocarbon-ion-implanted silicon wafer. We found that this novel silicon wafer has three unique gettering characteristics for the improvement of advanced CMOS image sensors.\(^{(25)}\) First, a hydrocarbon-molecular-ion-implanted silicon wafer has a high gettering capability for metallic impurities.\(^{(26)}\) Second, with respect to the oxygen out-diffusion barrier effect, this wafer can control oxygen out-diffusion to the device active region from the CZ-grown silicon substrate during the device fabrication process.\(^{(27)}\) Third, it produces a hydrogen passivation effect for process-induced defects (interface state defects) in CMOS image sensors. The hydrogen was gettered in the hydrocarbon molecular ion projection range.\(^{(28)}\) We expected that the hydrogen would out-diffuse to the device active region from the projection range during the CMOS device fabrication process.\(^{(29)}\) This phenomenon will have a passivation effect on process-induced defects such as those in the Si/SiO\(_2\) interface states.\(^{(30)}\)
In this review article, the main objective is to demonstrate the metal, oxygen, and hydrogen impurity gettering capability of the hydrocarbon-molecular-ion-implanted silicon wafer using the CMOS image sensor fabrication process. Furthermore, we investigate the effect of white spot defects on the CMOS image sensor with and without hydrocarbon molecular ion implantation for the first time. Few previous studies of white spot defects were evaluated by CMOS image sensors using the proximity gettering technique. Therefore, we demonstrate that the novel silicon wafer has a high gettering capability and substantially decreases the white spot defect density as evaluated by dark current spectroscopy. It is extremely important to understand the gettering properties of the hydrocarbon-molecular-ion-implanted silicon wafer for understanding advanced CMOS image sensors from the viewpoints of fundamental material science and silicon wafer gettering design for CMOS image sensor fabrication.

We propose a solution to CMOS image sensor technical issues for designing silicon wafer gettering using hydrocarbon molecular ion implantation to achieve a high performance in CMOS image sensor manufacturing.

2. Materials and Methods

2.1 Production concept of hydrocarbon-molecular-ion-implanted silicon wafer

Figure 2 shows the production concept of hydrocarbon-molecular-ion-implanted epitaxial silicon wafers. First, the hydrocarbon-molecular-ion-implanted silicon substrate top surface
without an oxidation film and additional recrystallization heat treatment using a hydrocarbon molecular ion implanter (model CLARIS) were provided by Nissin Ion Equipment, Japan.\(^{(31-33)}\) The hydrocarbon molecular ion forms a carbon and hydrogen projection range after implantation. Second, the epitaxial silicon layer was deposited on the silicon substrate surface by chemical vapor deposition. Thus, the production concept of this novel silicon wafer is extremely simple.\(^{(24)}\)

Figure 3 shows the SIMS depth profile measured on hydrocarbon-molecular-ion-implanted epitaxial silicon wafers after heat treatment. The hydrocarbon molecular ion implantation projection range forms carbon and hydrogen implantation elements. In this projection range, the carbon peak concentration is \(2 \times 10^{20} \text{cm}^{-3}\) and the hydrogen peak concentration is \(8 \times 10^{20} \text{cm}^{-3}\). Moreover, the carbon concentration in this projection range is three orders of magnitude higher than the solid solubility of carbon impurities in the silicon crystal after heat treatment.

### 2.2 Sample preparation and evaluation techniques

The experimental samples used in this study were (100) phosphorus-doped CZ silicon single crystals of 750 μm thickness. Their resistivity was 10 Ωcm and their oxygen concentration was \(1.4 \times 10^8 \text{atoms/cm}^3\) (old ASTM). The sample wafers were subjected to \(\text{C}_3\text{H}_5\) hydrogen molecular ion implantation of 80 keV to a dose of \(1 \times 10^{15} \text{cm}^2\). Their resistivity was also 10 Ωcm and the epitaxial layer grown by chemical vapor deposition was 5 μm in thickness. The metallic impurity contaminated hydrocarbon-ion implanted epitaxial silicon wafers were fabricated by surface spin coating with a metallic-impurity-contaminated SC1 and SC2 acid solution. The nickel and copper surface metallic impurities contamination concentration at a level of \(1 \times 10^{13} \text{atoms/cm}^2\) on their surfaces. After the metallic impurity contamination, the wafers were annealed by heat treatment. The metal, oxygen, and hydrogen impurities of diffusion and the depth profile in the \(\text{C}_3\text{H}_5\) molecular ion implantation projection range (\(\text{C}_3\text{H}_5\) ions at 80 keV to

![Fig. 3. (Color online) SIMS depth profile of hydrocarbon-molecular-ion-implanted epitaxial silicon wafers after epitaxial layer growth process. Modified from Kurita et al.\(^{(26)}\) Copyright (2016) The Japan Society of Applied Physics.](image-url)
a dose of $1 \times 10^{15} \text{ cm}^{-2}$) were measured by secondary ion mass spectroscopy (SIMS) and atom probe tomography (APT) depth analysis before and after the CMOS heat treatment. The APT tool is very efficient in obtaining a 3D impurity distribution in semiconductor materials with an atomic-scale spatial resolution by CAMECA’s LEAP 5000 atom probe microscopy.\(^{(34)}\)

### 3. Results

#### 3.1 Metallic impurity gettering capability after heat treatment

Figure 4 shows the nickel and copper metallic impurities gettered in the hydrocarbon molecular ion implantation projection range after intentional surface metal contamination and diffusion heat treatment. These results indicate that the metallic impurities diffused to the hydrocarbon molecular ion projection range during heat treatment. Thus, the metallic impurities were gettered in the hydrocarbon molecular ion projection range. This indicates that the novel silicon wafer has a high gettering capability for metallic impurities.\(^{(26,30)}\)

#### 3.2 Oxygen out-diffusion barrier effect during device heat treatment

Figure 5 shows the SIMS depth profiles of oxygen impurities measured on the epitaxial silicon wafers with and without hydrocarbon molecular ion implantation after device heat treatment. The oxygen impurities out-diffused to the silicon epitaxial layer from the CZ-silicon grown substrate during the device heat treatment. For a hydrocarbon-molecular-ion-implanted epitaxial silicon wafer, the oxygen impurities were gettered in the hydrocarbon ion implantation projection range during the device heat treatment. The hydrocarbon ion implantation projection range proved to be an effective gettering sink for controlling out-diffused oxygen impurities from a CZ-grown silicon substrate. This result indicates that the oxygen out-diffusion

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**Fig. 4.** (Color online) SIMS depth profile of hydrocarbon-molecular-ion-implanted epitaxial silicon wafer after metallic impurity diffusion heat treatment. This epitaxial silicon wafer sample was C\(_3\)H\(_5\) hydrocarbon-molecular-ion-implanted at 80 keV to a dose of $1 \times 10^{15} \text{ cm}^{-2}$. Modified from Kurita et al.\(^{(26)}\) Copyright (2016) The Japan Society of Applied Physics.
magnitude of a hydrocarbon-ion-implanted epitaxial silicon wafer is one order lower than that of an epitaxial silicon wafer without hydrocarbon molecular ion implantation. Thus, the hydrocarbon-ion-implanted epitaxial silicon wafer has a higher gettering capability for oxygen impurities than the epitaxial silicon wafer without hydrocarbon molecular ion implantation.

3.3 Hydrogen gettering effect during device heat treatment

The hydrocarbon molecular ion implantation forms a hydrogen and carbon impurity projection range in the silicon wafer top surface region as shown in Fig. 2. The position of each ion implantation element peak depends on the hydrocarbon mass number and acceleration ion implantation energy. The acceleration energy of the hydrogen element is 2 keV and that of the carbon element is 22 keV. Thus, peak position hydrogen forms a silicon wafer top surface region. Figure 6 shows the SIMS depth profile of the hydrogen implantation element in the projection range measured on a hydrocarbon-molecular-ion-implanted silicon wafer before epitaxial growth. Figure 7 shows the SIMS depth profile of hydrogen in the projection range of the hydrocarbon-molecular-ion-implanted silicon wafer after epitaxial growth. The hydrogen concentration substantially decreased in the projection range after epitaxial growth.

This is a very interesting experimental data, because hydrogen is generally a light element and extremely easily diffuses in the silicon crystal bulk during the heat treatment. Thus, it does not remain in the silicon crystal bulk after the heat treatment. However, for hydrocarbon molecular ion implantation, it remained in the cross of the hydrogen and carbon projection range after the epitaxial growth as shown in Fig. 3.

Why did hydrogen remain in the hydrocarbon ion implantation projection range? The reason is that the cross of the hydrogen and carbon projection range forms stress and strain in the projection range. Thus, hydrogen is gettered by this stress and strain after the epitaxial growth.

Fig. 5. (Color online) SIMS depth profiles of oxygen impurities in epitaxial silicon wafers with and without hydrocarbon molecular ion implantation after CMOS image sensor fabrication process. Modified from Kurita et al. (25) Copyright (2015) The Japan Society of Applied Physics.
This indicates that a hydrocarbon-ion-implanted epitaxial silicon wafer has a high gettering capability for hydrogen impurity after the heat treatment.

3.4 Gettering capability of hydrocarbon-molecular-ion-implanted epitaxial silicon wafer fabricated by CMOS image sensor using dark current spectroscopy

We consider the gettering capability of a hydrocarbon-molecular-ion-implanted epitaxial silicon wafer using a CMOS device fabrication line. We evaluated the white spot defect densities of CMOS image sensors with and without hydrocarbon molecular ion implantation using dark current spectroscopy (DCS). DCS is an extremely powerful metallic impurity contamination analysis tool for charge-coupled devices (CCDs) and CMOS image sensors. This technical principle behind it is very similar to that of deep-level transient spectroscopy (DLTS). It is well known that DLTS is an extremely useful metallic impurity contamination analysis tool and that it is used to measure transient capacitance temperature dependence in metal Schottky barrier-junction or pn-junction capacitor diodes. DLTS makes it possible to determine the concentration of metallic-impurity-related defects and the defect activation energy. It also makes it possible to capture a cross section in the silicon band gap. Thus, DLTS can identify metallic impurities in silicon crystal bulk. However, it cannot measure DLTS signals using conventional CMOS device structures such as MOS-FET. DLTS signal measurements can only be applied to Schottky barriers or pn-junction diodes. In contrast, DCS can measure metallic-impurity-related electrical signals using a CMOS device structure. Moreover, the metallic impurity detection limit of DCS is two or three orders of magnitude higher than that of DLTS. Thus, solid-state image sensor manufacturers have used DCS to analyze metallic impurity contamination in CCD and CMOS image sensors.

Figure 8 shows the DCS spectra of a CMOS image sensor with and without hydrocarbon molecular ion implantation after the device fabrication process. These spectra were measured from a CMOS image sensor using CMOS image sensor pixel architecture, which was
We found that the DCS spectra of the CMOS image sensors have three peaks in DCS signals. Peak1 in the distribution corresponds to the dark current induced by process-induced defects during the CMOS image sensor fabrication process. We also found that the DCS spectrum of a CMOS image sensor with hydrocarbon molecular ion implantation substantially decreased by two DCS signal peaks (Peaks 2 and 3) as shown in Fig. 9.\(^{(43)}\)

Domengie et al. demonstrated the effects of DCS signals in a CMOS image sensor intentionally contaminated with metallic impurities (W and Au) and used DLTS to evaluate them.\(^{(46)}\) They found that the DCS signal has three peaks. They measured W- and Au-related deep-level defects in the space charge region using DLTS. They understood that the DCS signal peaks were enhanced by metallic-impurity (W and Au)-related deep level defects in the silicon band gap. Russo and coworkers demonstrated the effects of the dark current of a CMOS image sensor intentionally contaminated with metallic impurities (Cr, V, Cu, Ni, Fe, Ti, Mo, W, Al, and Zn) and used DCS to evaluate them.\(^{(47,48)}\) Their results indicate that the Mo, W, Cu, and Ti impurities have greater effects on dark current generation than other metallic impurities. This is because, the Mo, W, Cu, and Ti impurities form deep-energy-level defects in the silicon band gap.\(^{(11)}\)

Therefore, our observation of two DCS signal peaks (Peaks 2 and 3) indicates that metallic impurities form deep-energy-level defects in the device active area (i.e., the photo diode space charge region). These defects can act as generation-recombination centers (G-R centers) in the silicon band gap. Thus, photo diode leakage current increased the number of G-R centers, which increased after metallic impurity contamination during the device fabrication process. DCS signals strongly correlate with photo-diode leakage current. We understand that the two DCS signal peaks originated from the observed metallic-impurity-related defects in the photo diode space charge region. Thus, we assume that the CMOS image sensor with hydrocarbon molecular ion implantation has a high gettering capability during the device fabrication process.

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![Fig. 8.](image_url) (Color online) White spot defect counts of CMOS image sensors with and without hydrocarbon molecular ion implantation after CMOS image sensor fabrication process.\(^{(43)}\)
3.5 TEM observation results of hydrocarbon-molecular-ion-implanted epitaxial silicon wafer before and after the CMOS image sensor fabrication process

Figure 10 shows the cross-sectional TEM observation results we obtained for the hydrocarbon-molecular-ion-implanted epitaxial silicon wafer after the CMOS device heat treatment. We found that there were hydrocarbon-molecular-ion-implantation-related defects in the hydrocarbon molecular ion implantation projection range. The defects were approximately 5 nm in size and $1 \times 10^{16} \text{cm}^{-3}$ in density. No secondary extended defects such as dislocations and dislocation loops were found in the hydrogen molecular ion implantation projection range as shown in Fig. 10.

3.6 APT observation results of hydrocarbon-molecular-ion-implanted epitaxial silicon wafer after the CMOS image sensor fabrication process

Figure 11 shows the cross-sectional observation results we obtained for an APT needle-shaped specimen prepared using field ion beams from a hydrocarbon-ion-implanted epitaxial silicon wafer after the CMOS image sensor fabrication process. The capping material nickel was used to form the surface position. After preparing a needle-shaped specimen, we measured the three-dimensional impurity distribution in the hydrocarbon molecular ion implantation projection range after the CMOS image sensor fabrication process. Figure 12 shows the APT map of a hydrogen-molecular-ion-implanted epitaxial silicon wafer in the hydrogen molecular ion implantation projection range after the CMOS image sensor fabrication process. The map data indicates that the carbon atoms agglomerate themselves in various forms such as carbon complexes during the CMOS heat treatment. Moreover, oxygen atoms segregated carbon complexes during the CMOS heat treatment.
4. Discussion

4.1 Gettering mechanism of hydrocarbon-molecular-ion-implanted epitaxial silicon wafer

Why does the CMOS image sensor with hydrocarbon molecular ion implantation has a high gettering capability during the device fabrication process? There are two possible gettering mechanisms that can explain this.
4.1.1 Relaxation-induced gettering mechanism\(^{(50,51)}\)

Metallic impurities can be gettered by ion-implantation-related defects such as dislocations and dislocation loops. A previous study showed high-energy ion-implantation-induced defects in the projection range. These defects induced strain and stress in the silicon matrix interface. Their effectiveness in gettering impurities is explained by the Cottrell effect, in which the solubility of an impurity atom is greater in the vicinity of secondary extended defects such as dislocations and dislocation loops.

The experimental results we obtained from the TEM observation are shown in Fig. 10. The obtained TEM observation results did not reveal any secondary extended defects in the hydrocarbon molecular ion implantation projection range. However, our previous study results showed that a hydrocarbon-molecular-ion-implanted epitaxial silicon wafer can getter metallic impurities after device heat treatment as shown in Fig. 4. Therefore, the relaxation-induced gettering mechanism cannot account for our experimental results.

4.1.2 Segregate-induced gettering mechanism\(^{(52)}\)

With the segregate-induced gettering mechanism, the solid solubility of metallic impurities in gettering sinks increased substantially more than that for silicon crystal bulk without gettering sinks. Figure 4 shows the SIMS depth profile of metallic impurities in the hydrocarbon molecular ion projection range after metallic impurity diffusion annealing. The metallic impurities are gettered by the hydrocarbon molecular ion projection range. The metallic impurity concentration in the projection range is higher than the solid solubility in silicon crystal bulk. Thus, the experimental results indicate that the hydrocarbon molecular ion projection range forms segregate-induced gettering sinks in silicon crystal bulk.

4.2 Origin of gettering sinks in hydrocarbon-molecular-ion-implanted epitaxial silicon wafer

What is the origin of metallic impurity gettering sinks in the hydrocarbon molecular ion implantation projection range? We used APT to observe the hydrogen molecular ion implantation-related defects in the implantation projection range (Fig. 12).\(^{(43)}\) The obtained results indicate that the defects form carbon oxygen agglomerates such as carbon complexes in the implantation projection range after the CMOS image sensor fabrication process. We analyzed the APT map of carbon complexes in determining the hydrogen molecular ion implantation range of the carbon complex density and size using Integrated Visualization and Analysis Software (IVAS) from CAMECA.\(^{(53)}\) We determined that the carbon complex density is \(1 \times 10^{16} \text{ cm}^{-3}\) and that the size is 5 nm. This means that we observed the same hydrogen molecular ion implantation-related defects such as carbon complexes in the hydrogen molecular ion implantation projection range that we did using TEM observation results after the CMOS image sensor fabrication process (see Fig. 10).
Why are the gettering sinks of carbon complexes in the ion implantation projection range effective for metallic impurity gettering? We assume that these sinks strongly interact with metallic impurities in a hydrocarbon molecular ion implantation projection range. Shirasawa et al. considered the above issues from the viewpoint of theoretical solid-state physics.\(^{(54–56)}\) They obtained theoretical results by first-principles calculation to determine the binding energies of metallic impurities and their complexes with intrinsic point defects, which were found to be the cause of the effectiveness of hydrogen molecular ion implantation gettering sinks. They indicated that the origin of the metallic impurity gettering sink must be interstitial carbon (Ci) and intrinsic point-defect complexes (carbon self-interstitial clusters, vacancy-oxygen pairs, and vacancy-hydrogen pairs).

Furthermore, Okuyama et al. demonstrated that the gettering behavior of hydrogen in the hydrocarbon molecular ion implantation projection range after epitaxial growth can be calculated by technology computer-aided design with a reaction model in which hydrogen binds to a carbon and silicon self-interstitial cluster (Cs-I).\(^{(57)}\) This calculation result indicates that a Cs-I cluster is an extremely effective gettering sink for metallic impurities in the hydrocarbon molecular ion implantation projection range.

Therefore, we propose that the hydrocarbon molecular ion implantation projection range must be formed in configurations such as carbon complexes, vacancy-oxygen pairs, and vacancy-hydrogen pairs for effective gettering sinks. We also used APT to observe the hydrocarbon molecular ion implantation projection range. The results we obtained support the hypothesis that the observed carbon complexes are one of the origins of gettering sinks in the hydrocarbon molecular ion implantation projection range after the CMOS device fabrication process. We believe that the segregated-induced gettering sinks originate from carbon complexes in hydrocarbon molecular ion implantation projection in silicon wafers.

5. Conclusions

The advanced CMOS image sensor fabrication process involves a number of serious technical issues such as metallic impurity contamination, dark current, and image lag. We consider that these issues require silicon wafer technology solutions using hydrocarbon molecular ion implantation. Accordingly, we developed a hydrocarbon-molecular-ion-implanted epitaxial silicon wafer with a high gettering capability for metal, oxygen, and hydrogen impurities during the CMOS image sensor fabrication process. This novel silicon wafer substantially decreased the white spot defect density and dark current in the CMOS image sensor fabrication line. In conclusion, we believe that such a wafer will be beneficial for advanced CMOS image sensor manufacturing.

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