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Vibrational exciton nanoimaging of phases and domains in porphyrin nanocrystals

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Significance

Molecular coupling defines fundamental properties of materials, yet spectroscopic access and imaging have remained challenging due to the short length scales of order and disorder and the low energy scales of interactions. We employ vibrational excitons as local structural probes in nanoscale infrared imaging and spectroscopy to map molecular order in a model organic electronic material (octaethylporphyrin ruthenium(II)carbonyl (RuOEP)). We observe coexistence of both amorphous and crystalline phases. Even for highly crystalline porphyrin, the size of individual ordered regions can remain limited to a few molecular lengths. This approach of vibrational exciton nanospectroscopy is broadly applicable to many interacting molecular systems and can provide insight into structure and energy transfer in organic electronics, proteins, or other biological complexes.

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The authors declare no competing interest.

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Data deposition: Data used for this paper have been deposited in the Open Science Framework (OSF) repository, https://osf.io/72vcx/.

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RuOEP aggregates were formed by phase segregation in a thin-film molecule polymer blend with a 1:10 mass ratio of regioregular poly(3-hexylthiophene-2,5-diyl) (P3HT) (Fig. 1B), with morphology and crystallinity controlled by solvent annealing in chloroform vapor for variable duration from 0 to 240 min (31). We monitored film quality using a combination of far-field Fourier transform infrared (FTIR), atomic force microscopy (AFM), SAXS, and synchrotron infrared nanospectroscopy (SINS) (Advanced Light Source, beamline 5.4) (25), with a representative SINS spectrum shown in Fig. 1C. We measured the nanoscale spectroscopic response from RuOEP aggregates using IR s-SNOM, as shown in Fig. 1A, based on a low-noise actively and passively stabilized tunable femtosecond infrared (mid-IR) laser source (HarmoniXX difference frequency generation [DFG], APE; Levante optical parametric oscillator [OPO], APE; Flint, Light Conversion), tuned to the frequency of the RuOEP carbonyl stretch at ~1,930 cm⁻¹, optimized with ~100-cm⁻¹ bandwidth for maximal spectral irradiance and sensitivity. As illustrated in Fig. 1D, we measured spatio-spectral images of both crystalline and noncrystalline regions within the RuOEP aggregates as spectrally resolved voxels with grid spacings of 20 to 50 nm. The tip-scattered near-field signal was measured interferometrically as amplitude |A_{NF}(\vec{\nu})| and phase \Phi_{NF}(\vec{\nu}) spectra with 2-cm⁻¹ spectral resolution by using lock-in demodulation at the second harmonic of the cantilever motion (28, 33) (for details, see Materials and Methods).

**Results**

**RuOEP Aggregate Formation and Growth.** We first measured the ensemble-averaged vibrational response of RuOEP in P3HT thin films as a function of chloroform solvent annealing time using FTIR reflectance spectroscopy. Prior to solvent annealing, thin film samples exhibited a broad carbonyl stretch ν₀ centered at 1,931 cm⁻¹ (Fig. 2A, blue). That peak then narrowed from an initial full-width at half-maximum of \Gamma(\nu₀) = 34 ± 4 cm⁻¹ (Fig. 2A, blue) to \Gamma(\nu₀) = 11 ± 1 cm⁻¹ within the first 20-min

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annealing time (Fig. 2A, orange). With further annealing, the \( \nu_0 \) intensity decreased as two satellite peaks \( \nu_- \) and \( \nu_+ \) (Fig. 2A, red) emerged. Both the larger peak, \( \nu_- \approx 1\,920 \text{ cm}^{-1} \), and the smaller peak at \( \nu_+ \approx 1\,947 \text{ cm}^{-1} \) were spectrally narrow with \( \Gamma(\nu_\pm) \approx 8 \text{ to } 10 \text{ cm}^{-1} \), after 60-min vapor exposure (Fig. 2A, red).

Based on ensemble-averaged FTIR measurements, \( \bar{\nu}_0 \) was previously assigned to isolated RuOEP molecules within the P3HT matrix, while \( \nu_- \) and \( \nu_+ \) were assigned to crystalline RuOEP phases separated from the P3HT (31). As shown in Fig. 2B, transition dipole coupling of the metal-carbonyl groups in the crystalline phase resulted in a splitting of the carbonyl stretch mode \( \bar{\nu}_0 \) leading to the distinct modes \( \nu_- \) and \( \nu_+ \).

For the unannealed film, as shown in the AFM height image in Fig. 2D, we observed a uniformly disordered topography with small height variations of \( \pm 7 \pm 1 \text{ nm} \). RuOEP aggregates formed already after 10 min of solvent annealing, with typical aggregate sizes of \( \sim 100 \text{ to } 200 \text{ nm} \), yet amorphous morphology (Fig. 2E). After 150-min solvent annealing, these aggregates transitioned into uniform square crystals with lateral dimensions of 200 to 500 nm and identifiable crystal faces, indicative of the emergent monoclinic crystal habit (34), as seen in Fig. 2F.

In complementary SAXS measurements of crystallinity, a broad peak appeared after 20-min annealing. Fits to the Scherrer equation revealed a coherence length of only \( 1.9 \pm 0.3 \text{ nm} \), indicating short-range order on the two to three molecular lengths scale. With longer annealing times, the emergence of a narrow peak at the same scattering angle with continuous presence of the broad peak indicated growth of a crystalline phase with coherence length reaching \( \sim 70 \pm 10 \text{ nm} \) for the longest annealing times of 180 min.

We next used IR s-SNOM to measure the spectral response from nanoscale regions within an individual RuOEP aggregate, first at early annealing times (Fig. 2E). Fig. 2C shows a representative IR s-SNOM spectrum \( \Phi(\bar{\nu}_\nu) \) within an RuOEP aggregate solvent annealed for 10 min (green), revealing a single broad peak at \( 1\,931 \text{ cm}^{-1} \) with a width of \( \sim 14 \text{ cm}^{-1} \). Spectra for different aggregates, all annealed for the same time, were similar, with only some \( (\pm 5 \text{ cm}^{-1}) \) variation in line width between aggregates. While previous far-field studies have attributed this spectrally broad feature to isolated RuOEP molecules within the P3HT matrix (31), IR s-SNOM instead shows that this spectral feature must be assigned to already phase-segregated RuOEP aggregates.

With further annealing, two additional peaks \( \bar{\nu}_- \) and \( \bar{\nu}_+ \) appeared in the near-field spectra, along with \( \bar{\nu}_0 \), as shown in Fig. 2C (red), indicating phase coexistence of both a strongly coupled crystalline RuOEP alongside weakly interacting RuOEP in the same nanoscopic volumes (\( \sim 1 \times 10^7 \text{ molecules or } \sim 0.02 \text{ amol} \)).

**Nanoinaging of Partially Ordered Aggregates.** We next mapped the nanoscale spatial variations within RuOEP aggregates through changes in spectral broadening. Fig. 3A shows a map of \( \Gamma(\bar{\nu}_0) \) derived from fits of the voxel array across a RuOEP aggregate after a 60-min solvent anneal, with two representative spectra and fits shown in Fig. 3B. We found narrowed line widths \( \Gamma(\bar{\nu}_0) \) ranging from 10 to 14 cm\(^{-1}\) across this aggregate and typical for this annealing time. However, distinct nanoscale regions of broad and narrow line widths exist within these aggregates (Fig. 3A, voxels, and Fig. 3C, surface plot), which we attributed to spatial variations in inhomogeneous broadening.

From the IR s-SNOM spectra \( \Phi(\bar{\nu}_\nu) \), we then derived the fraction of molecules in the crystalline phase \( n(\bar{\nu}_-) \) and non-crystalline phase \( n(\bar{\nu}_0) \). We started by determining the relative intensities \( \Phi(\bar{\nu}_-) \) and \( \Phi(\bar{\nu}_+) \), proportional to \( n_0 / n_j^2 \) for a phase \( j \), from fits to \( \Phi(\bar{\nu}_\nu) \) spectra using the point-dipole model (details are in SI Appendix). Under the approximation of uniform transition dipole moment strength of the metal-carbonyl peak in both phases (35), we can define a crystallinity index \( \chi_{\text{RuOEP}} \):

\[
\chi_{\text{RuOEP}} = n(\bar{\nu}_-)/[n(\bar{\nu}_-)+n(\bar{\nu}_0)] \approx \Phi(\bar{\nu}_-)/[\Phi(\bar{\nu}_-)+\Phi(\bar{\nu}_0)].
\]

\( \chi_{\text{RuOEP}} \) is independent of molecular orientation (29) or small variations in tip sample coupling as a result of the isotropic transition dipole moment of \( \nu_- \) and \( \nu_0 \) in both phases.

We display spatially resolved nanoscale variation in crystallinity \( \chi_{\text{RuOEP}} \) in Fig. 3D, with its interpolated three-dimensional (3D) representation in Fig. 3F. We show two spectra representative of the variation in crystallinity in Fig. 3E, which were selected to highlight two voxels with nearly identical response of \( \bar{\nu}_- \) \((V_1, \text{ red}; V_2, \text{ blue}) \), yet a stronger signal from the \( \bar{\nu}_0 \) mode in the \( V_2 \) spectrum, which we attributed to a lower \( \chi_{\text{RuOEP}} \). As can be seen, the aggregate is characterized by a generally low crystallinity \( \chi_{\text{RuOEP}} \sim 0.4 \), with a domain of much higher crystallinity (Fig. 3F).
higher crystallinity \( X_{\text{RuOEP}} \approx 0.75 \) (left center). We note the associated anticorrelation of \( X_{\text{RuOEP}} \) (Fig. 3F) compared with \( \Gamma (\nu_0) \) (Fig. 3C), as expected for a decrease of inhomogeneous broadening with increasing crystallinity.

We next imaged an RuOEP aggregate with high crystallinity. Nanoscale voxel array measured with 20-nm grid spacing (Fig. 4A) and corresponding surface plot (Fig. 4C) revealed a greater overall crystallinity \( X_{\text{RuOEP}} \approx 0.9 \) across most of the lower right region, with somewhat lower values of \( \approx 0.75 \) to 0.85 across much of the upper right and the lower left. We observed variation in crystallinity across the imaged region with spatial correlation on the \( \approx 100\) nm length scale, well resolved within both the spatial resolution of \( s\)-SNOM determined by the \( \approx 20\) nm tip radius and the 20-nm voxel spacing.

Intermolecular Coupling in Crystalline Regions. We then analyzed for variations in peak position associated with crystallinity. Fig. 5A shows representative IR \( s\)-SNOM spectra in the \( \nu_- \) and \( \nu_0 \) spectral region, with corresponding fits for a 60-min solvent-annealed sample. In regions with low crystallinity \( X_{\text{RuOEP}} = 0.28 \) (blue), the peak position of \( \nu_- \) appeared near 1,924 cm\(^{-1}\). In regions with higher crystallinity \( X_{\text{RuOEP}} = 0.61 \) (green), \( X_{\text{RuOEP}} = 0.74 \) (orange), and \( X_{\text{RuOEP}} = 0.93 \) (red), we observed an overall red shift in the peak position with increasing crystallinity.

Fig. 5B shows the corresponding 3D histogram of the variation in center frequency of \( \nu_- \) versus \( X_{\text{RuOEP}} \) from statistical analysis of fits to voxel arrays from seven different nanocrystals with moderate to high crystallinity index \( X_{\text{RuOEP}} = 0.40 - 0.99 \). The 90% probability contours from four

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**Fig. 4.** (A) Nanoscale map of \( X_{\text{RuOEP}} \) measured with 20-nm grid spacing for a RuOEP aggregate with high crystallinity. (B) Representative spectra and fits for locations indicated in A. (C) Corresponding 3D surface plot of \( X_{\text{RuOEP}} \) over the region in A.

**Fig. 5.** (A) IR \( s\)-SNOM spectra measured at four locations within RuOEP nanocrystals showing spectral shifts of \( \nu_- \). (B) Histogram of \( \nu_- \) peak center versus \( X_{\text{RuOEP}} \) for voxel arrays from multiple RuOEP nanocrystals within the same sample. Max, maximum; min, minimum. (C) Vibrational exciton model calculation with splitting of \( \nu_- \) and \( \nu_0 \) as a function of aggregate size for several models: 1D\( _{\text{conf}} \) (blue), 3D\( _{\text{RuOEP}} \) (red), 3D\( _{\text{local}} \) (green), and 1D\( _{\text{w}} \) (black). (D) Nanomapping of crystalline domain size. Overlay shows delocalization length as determined by spectral shifts of the vibrational exciton spectra, measured with 20-nm grid spacing, with topography shown in the background. (E) Schematic of phase segregation during solvent annealing. Initially, seed crystals form and increase in size, followed by Ostwald ripening and associated increase in crystallinity. Dark blue circles indicate amorphous porphyrin; light blue through red circles indicate the size of crystalline domains.
selected nanocrystals were overlaid (dashed lines), along with the results from the two individual fits from Fig. 5A (open circles). A correlation of red shift of $\nu$ with increasing crystallinity was observed. Table 1 summarizes the general range of line shape parameters for these and other aggregates of different crystallinity with experimental values for $\bar{\nu}$ as $1,917$ to $1,924$ cm$^{-1}$. According to models with high crystalline fraction $X_{\text{RuOEP}} \geq 0.8$, the $1,924$ cm$^{-1}$ and $1,920$ cm$^{-1}$ we identified the maximum peak shift with $\bar{\nu}$ of $1,917$ cm$^{-1}$ for highly ordered nanocrystals annealed for 180 min and the minimum peak shift with $\bar{\nu}$ of $1,924$ cm$^{-1}$ from crystals with low $X_{\text{RuOEP}}$. The observed correlation between line shape and crystallinity thus identified nanoscale subensembles with variations in local environment, which we related to intermolecular coupling and possible finite size effects of the crystalline domains as described and modeled below.

**Vibrational Exciton Model.** In order to determine the length scale of molecular ordering from peak splitting and spectral shifts, we modeled intermolecular coupling of the carbonyl stretch as a vibrational Frenkel excitons (22, 36, 37). The dipole coupling energy $\tilde{V}_{m,n}$ between adjacent molecules with transition dipole moments $\mu_m$ and $\mu_n$ and intermolecular distance $\tilde{r}$ is given by

$$\tilde{V}_{m,n} = \frac{1}{4 \pi \epsilon_0 |\tilde{r}|} |\tilde{\mu}_m - \tilde{\mu}_n| - 3(|\tilde{\mu}_m \cdot \tilde{r}|)(|\tilde{\mu}_n \cdot \tilde{r}|)/|\tilde{r}|^2.$$  

We then calculated the new energy eigenstates, forming the vibrational exciton band to first order in perturbation theory from the sum over all intermolecular interactions, sensitive to the number of interacting molecules and relative molecular arrangement. Based upon typical values of $\bar{\mu} \sim 1$ Debye and intermolecular spacing of $\tilde{r} \sim 1$ nm, the intermolecular coupling was on the order of $\tilde{V}_{m,n,\pm 1} = 8$ cm$^{-1}$ (40, 41). This interaction strength meets the requirements of strong coupling as originally defined by Förster for $2 \tilde{V}_{m,n} > \Gamma_{\text{eff}}$ and leads to the formation of vibrational excitons, that manifests itself as a splitting of the metal-carbonyl resonance from the uncoupled values of $\bar{\nu} = 1,931$ cm$^{-1}$ into two peaks, $\bar{\nu}_-$ and $\bar{\nu}_+$ (37).

To model the evolution of $\bar{\nu}_-$ and $\bar{\nu}_+$ as a function of molecular ordering, with results summarized in Table 2 for different molecular arrangements, we first explored a simplified one-dimensional (1D) chain model with the carbonyl dipole moments $\mu_c$ aligned parallel with respect to the molecular chain. Nearest-neighbor interactions in this 1D model (1D$_{\text{NN}}$) gave rise to only a single infrared (IR) active mode $\bar{\nu}_-$, with frequency progressively shifting with increasing chain length $l$, asymptotically approaching $\nu^\text{IR} = 1,917$ cm$^{-1}$ for $l > 12$ nm (details are in SI Appendix). This shift of $\bar{\nu}_-$ as a function of $l$ can be empirically fit by using $\bar{\nu}_-(l) = (\bar{\nu}_- - \nu^\text{IR}) \times \exp(1/l) + \nu^\text{IR}$ with characteristic length scale $l_c = 1.6$ nm for a coupling constant of $\bar{V}_{m,n,\pm 1} = -7.1$ cm$^{-1}$. Extending this model to two molecules per unit cell, consistent with known structures of RuOEP polymorphs (40, 42), and tilting alternating carbonyl stretch $\mu_c$ in the chain by angles $\theta$ and $-\theta$, respectively, shifts spectral weight from $\bar{\nu}_-$ to $\bar{\nu}_+$ (1D$_{\text{Rot}}^\text{NN}$). Matching the experimentally observed spectral-intensity ratio $\bar{\nu}_-/\bar{\nu}_+$, the 1D$_{\text{Rot}}$ results in the same values for $\nu^\text{IR} = 1,917$ cm$^{-1}$ and $l_c = 1.6$ nm for $\theta = 22.5^\circ$ (Fig. 5C, blue). However, in contrast to experiments, this model does not yet reflect the observed asymmetric splitting between $\bar{\nu}_-$ and $\bar{\nu}_+$.

Although the exact crystal structure of our chloroform solvent- annealed porphyrin derivative nanocrystals is unknown, we can develop 3D exciton models for two closely related RuOEP polymorphs (40, 42) of known crystal structure to assess whether the 1D model already captures the length scale and energy shifts of intermolecular coupling associated with molecular order. Both structures have two molecules per unit cell, intermolecular spacing of 8 to 11 Å, and a mixture of positive and negative coupling constants, but differ in details of packing and molecular orientation within the unit cell. Using the triclinic structure determined by Salzmann et al. (42) and $\bar{V}_{m,n,\pm 1} = 8.5$ cm$^{-1}$ (3D$_{\text{triclinic}}$) results in a single IR active mode $\nu^\text{IR} = 1,917$ cm$^{-1}$, in good agreement with experiment, and $l_c = 0.7$ nm, but lacks an IR active $\nu_+$ mode (Fig. 5C, red), like 1D$_{\text{NN}}$. In contrast, the tetragonal structure from Miranda et al. (40) and $\bar{V}_{m,n,\pm 1} = -8.3$ cm$^{-1}$ results in a $\nu^\text{IR}$ mode at $1,917$ cm$^{-1}$, $\nu^\text{IR} = 1,960$ cm$^{-1}$, and $l_c = 2.3$ nm (3D$_{\text{tetragonal}}$). This splitting shows $\nu^\text{IR}$ asymmetrically split from the uncoupled frequency, but blue-shifted by 11 cm$^{-1}$ compared to experiment.

Comparing the 1D and 3D models, we conclude that $l_c$ of 12 nm and associated redshift of $\bar{\nu}_-$ is insensitive to the exact crystal structure, but the asymmetric splitting between $\nu^\text{IR}$ and $\nu^\text{IR}$ appears to depend on the details of the lattice structure, in particular, the next-nearest-neighbor interaction and its sign relative to the nearest-neighbor term. However, the characteristic length scale of the vibrational excitons shift of $\bar{\nu}_-$ being similar between 1D and 3D models suggests that an effective 1D model (1D$_{\text{eff}}$) can adequately describe the fundamental vibrational exciton formation. With two molecules per unit cell, relative tilt of $\mu_c$, and an additional next-nearest-neighbor term $\bar{V}_{m,n,\pm 2}$, as characteristic for the range of experimentally known crystal structures, an effective 1D model yields a blue shift of $\nu^\text{IR}$ and asymmetric splitting. Specifically, for imposing a small next-nearest-neighbor coupling term $\bar{V}_{m,n,\pm 2} = 1$ cm$^{-1}$, within the range of typical values for known 3D structures, this effective 1D model results in $l_c = 1.2$ nm with splitting $\nu^\text{IR} = 1,917$ cm$^{-1}$ and $\nu^\text{IR} = 1,949$ cm$^{-1}$, in good agreement with experimental observations (Fig. 5C, discrete black lines).

**Growth of Nanoscale Crystalline Regions.** The peak positions $\bar{\nu}_-$ and associated values of crystallinity $X_{\text{RuOEP}}$ (Fig. 5B) then

### Table 1. Range of vibrational line shape parameters of metal-carbonyl response within RuOEP nanocrystals as a function of crystallinity $X_{\text{RuOEP}}$ with the corresponding number of coupled molecules $N$ as derived from the vibrational exciton model

| $X_{\text{RuOEP}}$ | $\bar{\nu}_-$ | $\Gamma(\bar{\nu}_-)$ | $\bar{\nu}_0$ | $\Gamma(\bar{\nu}_0)$ | $\bar{\nu}_+$ | $\Gamma(\bar{\nu}_+)$ | $N^*$ | $l$ (nm) |
|-------------------|--------------|----------------------|--------------|----------------------|--------------|----------------------|-----|---------|
| 0%                | --           | --                   | 1,930 to 1,932 | 15 to 30              | --           | --                   | 1   | 0       |
| 40 to 70%         | 1,920 to 1,923 | 6 to 9               | 1,930 to 1,932 | 11 to 15             | 1,945 to 1,949 | 6 to 12              | 2 to 6 | 1 to 5     |
| >90%              | 1,917 to 1,921 | 5 to 10              | --           | --                   | 1,945 to 1,949 | 6 to 12              | >6  | >5       |
| 1D$_{\text{eff}}$ model$^*$| 1,917 | --                   | 1,931 | 1,949 | -- | --                   | $\infty$ | 1.2      |

$^*$ Spectroscopic parameters are in cm$^{-1}$.

$^*$ $N$ is the number of molecules based on 1D$_{\text{eff}}$ model with intermolecular spacing of 1 nm.

$^*$ Values $\nu^\text{IR}^\mp$, $\nu^\text{IR}^\pm$, and $l_c$ calculated for infinite chain with model parameters $\bar{\nu}_0 = 1,931$ cm$^{-1}$ (fixed).
Table 2. Coupling parameters $\tilde{V}_{m,m',\ell}$ and predicted values of $\nu^{\infty}$, $\nu^{\infty}$, $l_{c}$ for different models of RuOEP

| Model          | Experiment | $1D_{NN}$ | $1D_{trigonal}$ | $3D_{tetragonal}$ | $1D_{art}$ |
|----------------|------------|-----------|-----------------|-------------------|------------|
| $\tilde{V}_{m,m',\ell}$ (cm$^{-1}$) | —         | -7.1       | -7.1            | 8.5               | -8.3       | -8.0       |
| $\nu^{\infty}$ (cm$^{-1}$)         | 1,917      | 1,917      | 1,917           | 1,917             | 1,917      | 1,917      |
| $\nu^{\infty}$ (cm$^{-1}$)         | 1,949      | —          | 1,944           | —                 | 1,960      | 1,949      |
| $l_{c}$ (nm)          | —          | 1.6        | 1.6             | 0.7               | 2.3        | 1.2        |

Superscripts are as follows: NN, model with nearest-neighbor interactions; trigonal, based on structure by Miranda et al. (40); tetragonal, based on structure by Salzmann et al. (42).

This nanoscale- and molecular-level insight into the presence of disordered and ordered phases and vibrational excitons confined to few-nanometer domains can provide critical understanding for the design of functional materials. Both crystalline and amorphous phases may occur in devices made with porphyrins or other molecular materials, and tradeoffs exist between benefits of crystalline phases with typically higher charge carrier mobility and exciton diffusion length versus amorphous phases that may exhibit improved interfacial miscibility and improved charge separation (48–51). The spatial organization of domains remains poorly understood across multiple length scales and is often only characterized by ensemble-averaged measurements due to the combined challenge of short length scales and low-energy scales of interaction. X-ray nanomapping and transmission electron microscopy (TEM) methods are poorly suited to molecular materials due to low-scattering cross-sections and low-dose imaging. Even specialized low-dose imaging has yet been unable to distinguish adjacent amorphous and crystalline phases (52). While previous studies using TEM and far-field spectroscopy had assigned broad spectral features to isolated RuOEP with the P3HT matrix (31), we instead observed, using vibrational excitation nanospectroscopy, the coexistence of amorphous and crystalline phases, with domains as small as a few molecules. We found molecular crystals exhibiting both long-range order, as determined by sharp SAXS diffraction peaks, yet with amorphous phase material in the same sample. Albeit a near-surface probe, the IR s-SNOM spectroscopic depth of ∼20 nm is well suited to investigate a wide range of functional materials and organic electronics that commonly exhibit both few-nanometer crystalline domains and spatial heterogeneity on multiple length scales (53).

Spectroscopic measurements of vibrational mode coupling in molecular aggregates can add understanding of fundamental physical properties of vibrational excitons and their spatial delocalization. The vibrational exciton is a delocalized, collective vibration that exhibits phonon softening and increase of splitting with increasing delocalization. However, lattice disorder in heterogeneity in coupling strengths can lead to localization of the vibrational exciton, in addition to the confinement we model (16). Additionally, thermal fluctuations can lead to either self-trapping or Anderson localization, depending on temperature (12, 54). These localization phenomena remain broadly challenging, both theoretically and experimentally, and temperature-dependent IR s-SNOM combined with high-resolution structural measurements could add fundamental understanding of energy transport in coupled molecular systems.

In conclusion, we used IR s-SNOM spectroscopy of a vibrational exciton as a probe to image from local molecular order to nanoscale crystallinity. We measured coexistence of both ordered and disordered phases in RuOEP within the same morphological region. From spectroscopic signatures of
vibrational exciton delocalization, we observed that spatial heterogeneity extends down to molecular scales. We found molecular crystals with long-range order and sharp SAXS diffraction peaks, yet with nanometer-scale coexistence of amorphous and crystalline phases with domains as small as a few molecules. Our approach of nanospectroscopy of coupled vibrational modes is readily generalizable and can provide fundamental insight into few-molecule to long-range order that determines charge, thermal, and energy transport.

Materials and Methods
Sample Preparation. A 1:10 mass ratio blend of porphyrin–polymer was prepared and mixed in a 1 wt% solution of 1.5 mL of chloroform and given 24 h to dissolve following sonication. Commercial PFHT and RuOEP were acquired from Rieke Metals Inc. and Sigma-Aldrich, respectively. Thin films were prepared by spin-coating from solution at 3000 rpm for 1 min onto freshly prepared, template-striped Au substrates. The thin films were subsequently solvent-anealed in chloroform vapor for up to 240 min.

Far-Field FTIR and SAXS. Far-field reflectance data were acquired by using a commercial FTIR spectrometer (Vertex v70, Bruker Optics) coupled to an IR microscope (Hyperion; Bruker Optics) and a water-cooled SiC global infrared (IR) microscope (Hyperion; Bruker Optics) and a water-cooled SiC global IR microscope (Hyperion; Bruker Optics) operating in intermittent contact (tapping) mode (56). The near-field tip-scattered signal was detected interferometrically in a backscattering geometry by an HgCdTe detector (MCT KLD-0.5-J1/DC11, Kolmar Technologies) using lock-in demodulation (Zurich Instruments HF2LI) at the second harmonic of the cantilever motion. Complex valued Fourier transform near-field spectra were represented as amplitude $A(\nu)$ and phase $\Phi(\nu)$, containing information about the dispersive and absorptive components of the IR response, respectively (28, 33). IR s-SNOM spectra were collected with 2 cm$^{-1}$ spectral resolution by using rapid-scan detection and averaging times of 1 to 4 min.

Data Availability. The procedures of experiments and simulations are described in detail in Materials and Methods and in SI Appendix. Data, computer code for fitting, and modeling code are available through the Open Science Framework (https://osf.io/72vwx) (57).

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