Diodes with Breakdown Voltages Enhanced by the Metal-Insulator Transition of \( \text{LaAlO}_3\)-\( \text{SrTiO}_3 \) Interfaces

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As discovered by A. Ohtomo and H. Y. Hwang, the interface between the TiO\(_2\)-terminated (001) surface of \( \text{SrTiO}_3 \) and \( \text{LaAlO}_3 \) can generate a conducting electron system \( \text{[1]} \). This system is an electron liquid \( \text{[2]} \), characterised by a carrier density of several \( 10^{13} \text{ cm}^{-2} \) and a room temperature sheet resistance in the range of \( 10^4 \Omega/\square \). For \( \text{LaAlO}_3 \) layers that are 3 or 4 unit cells (uc) thick, the electron system is close to a metal-insulator transition \( \text{[3, 4]} \). In \( \text{LaAlO}_3/\text{SrTiO}_3 \) heterostructures with a \( \text{LaAlO}_3 \) layer of at most 3 uc, the as-grown interface is insulating. However, an interface to a 3 uc thick \( \text{LaAlO}_3 \) layer becomes conducting if its carrier density is enhanced by applying a sufficiently large electric field in a field-effect transistor configuration. Correspondingly, the conductivity of an interface in a heterostructure with a 4 uc thick \( \text{LaAlO}_3 \) layer is switched off if the interface is depleted by a gate field.

The possibility to drive the interface through a metal-insulator transition by applying modest gate voltages is of interest for device applications. First, the abrupt change of the interface properties at the phase transition provides the possibility to operate devices with small input voltage swings. Second, controlling the presence of a metallic layer by switching an interface between the metallic and insulating ground state allows to effectively reconfigure a device. Here, we report on diodes in which the latter effect is utilised to raise breakdown fields and achieve large voltage-induced capacitance changes.

The principle of these devices is sketched in Fig. 1, which shows a cross sectional cut through a selfconducting diode based on a \( \text{SrTiO}_3 \) ((001), TiO\(_2\)-terminated) - \( \text{LaAlO}_3 \) (4 uc) - Au heterostructure. One contact to the diode is provided by a gold layer on top of the \( \text{LaAlO}_3 \), the other contact is given by a Au-plug that fills an ion-milled hole at the side of the device. This plug contacts the interface. In the devices fabricated the two contacts are spaced by a lateral distance of \( d \approx 1 - 30 \mu \text{m} \). If operated with a positive voltage applied to the top contact (Fig. 1a), the devices resemble Schottky diodes biased in forward direction, with a current flowing by tunnelling or thermal activation across the 4 uc thick (\( \approx 1.6 \text{ nm} \)) \( \text{LaAlO}_3 \) layer. With a negative voltage at the top contact (Fig. 1b), the devices are biased in reverse direction and, for voltages exceeding \( \approx 1 \text{ V} \), the metal-insulator transition is induced. At the transition the lower electrode of the diode therefore disappears. Thus, by applying such a voltage the effective length of the insulator in the device is enhanced from 1.6 nm to \( \approx 1 - 30 \mu \text{m} \). The diodes are therefore expected to sustain high reverse breakdown fields while supporting large forward currents.

Diodes with 3 uc \( \text{LaAlO}_3 \) layers operate according to the same principle, the main differences being that a finite voltage in forward direction is required to switch the conducting interface on, and that due to the thinner \( \text{LaAlO}_3 \) larger forward current densities can be obtained.
voltages exceed the measurement limit of 200 V at 300 K. As several 10 V. In some cases the reverse breakdown volt-
ages are stable and are rectifying up to the highest temperature applied. With increasing temperature the forward current sets in at reduced voltages and the breakdown voltage in reverse direction is lowered, reaching \( \approx 2 V \) at 273 C , in comparison to breakdown voltages of several 10 V at room temperature. The enhanced conduction is attributed to thermally excited charge carriers and thermally activated hopping. It is pointed out that at high temperatures the \( I(V) \) characteristics do not broaden significantly. In which manner this behavior is caused by the interface electron system, the LaAlO\(_3\) barrier, or the contacts remains to be analyzed.

To explore these ideas, we fabricated and experimentally investigated diodes of both types. The devices use epitaxial LaAlO\(_3\) films grown by pulsed laser deposition with reflective high energy electron diffraction as described in [3]. The LaAlO\(_3\) layers were deposited at 780 C in an oxygen pressure of \( 7 \times 10^{-5} \) mbar on TiO\(_2\)-terminated SrTiO\(_3\) substrates \([3, 4]\) with a subsequent cooldown to 300 K in 0.5 bar of oxygen \([3]\). The devices were patterned into the geometry shown in Fig. 1 by using the technique described in [7]. The gold top contacts were grown \textit{ex situ} using sputter deposition, then patterned by lift-off. The contacts to the interface were provided by Ar-ion milling photolithographically defined holes which were backfilled with sputtered gold. To avoid photoconductivity, the measurements were done after keeping the samples in dark for 24 hours.

The current-voltage \( (I(V)) \) characteristics of the devices are shown in Fig. 2. The characteristics are stable over weeks. As expected, the \( I(V) \) curves are highly asymmetric. In forward direction the devices have conductivities of \( \approx 10^{-3} \) \( (\Omega \text{cm})^{-1} \) as related to the top contact size. The \( I(V) \) characteristics show a hystere-
sis which we attribute to filling of trap states. In reverse direction the diodes show breakdown voltages of several 10 V. In some cases the reverse breakdown voltages exceed the measurement limit of 200 V at 300 K. As expected, devices with shorter \( d \) have smaller switch-on voltages (forward direction).

The size of the breakdown voltage provides clear evidence that the operation of the devices relies on the field-induced metal-insulator transition. If this was not the case the \( < 2 \text{ nm} \) thick LaAlO\(_3\) films would not sustain voltages \( > 200 \text{ V} \). The corresponding hypothetical elec-
tric fields of order \( 10^{11} \text{ V/m} \) exceed the breakdown field strength of any insulator by orders of magnitude.

To study the temperature dependence of the \( I(V) \) characteristics, the devices were heated to \( \approx 270 \text{ C} \), the temperature being limited by the diffusion of the gold contacts. To not reduce the SrTiO\(_3\) the heating was done in 0.5 bar of O\(_2\). As shown by Fig. 3 the diodes are stable and are rectifying up to the highest temperature applied. With increasing temperature the forward current sets in at reduced voltages and the breakdown voltage in reverse direction is lowered, reaching \( \approx 2 \text{ V} \) at 273 C , in comparison to breakdown voltages of several 10 V at room temperature. The enhanced conduction is attributed to thermally excited charge carriers and thermally activated hopping. It is pointed out that at high temperatures the \( I(V) \) characteristics do not broaden significantly. In which manner this behavior is caused by the interface electron system, the LaAlO\(_3\) barrier, or the contacts remains to be analyzed.

Because the lower electrode of the device is essentially removed when the devices are switched with the applied voltage \( V \) through the metal-insulator transition, the devices are expected to display a large change of capacitance \( C \) with \( V \). Fig. 4 shows the results of corresponding \( C(V) \) measurements. The capacitances of devices with 3 and 4 uc of LaAlO\(_3\) were measured as a function of applied dc-voltage, using a small ac-probe voltage (40 mV, 111 Hz, HP 4284A \textit{LCR-meter}). As expected, the capacitance of the 4 uc diodes (Fig. 4a) is reduced strongly at bias voltages between 0 V and \(-1 \text{ V} \), from \( \approx 1.35 \mu \text{F/cm}^2 \) to \( \approx 0.009 \mu \text{F/cm}^2 \). Also as expected, the capacity of the self-conducting diodes scales with the area of the top contact (Fig. 4b). Correspondingly, the small capacity of the unbiased, self-insulating diodes (Fig. 4a) is increased if the diodes are biased in forward direction. Within the accessible voltage range, however, these diodes do not reach the capacities of the unbiased 4 uc diodes. Similar to the \( I(V) \) characteristics, also the \( C(V) \) characteristics
FIG. 4. Capacitances of devices ($d = 10\, \mu m$) with 3 and 4 uc of LaAlO$_3$ measured at 300 K as a function of applied voltage.

show a hysteresis, again being presumably caused by filling of trap states. It is pointed out that in most devices the capacitance is smaller than the textbook value

$$C = \varepsilon_0\varepsilon_r \frac{A}{z}, \quad (1)$$

where $\varepsilon_0$ and $\varepsilon_r$ are the dielectric constant of vacuum and bulk LaAlO$_3$, respectively, $A$ is the area of the top contact and $z$ the thickness of the LaAlO$_3$ film. While most of the difference between the measured value and the one expected from Eq. 1 is likely caused by interface states which are in particular present at the Au-LaAlO$_3$ contact, it is pointed out that we do not presume Eq. 1 to accurately describe the capacity as besides the geometrical capacity further contributions are expected to add to the total capacity [8] (see also [9]).

In summary, taking advantage of the metal-insulator transition of LaAlO$_3$-SrTiO$_3$ interfaces, oxide diodes with large breakdown-voltages and capacities that are highly voltage-tunable have been fabricated. The devices are robust and can be operated far above room temperature.

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