Transition metal doped Sb@SnO$_2$ nanoparticles for photochemical and electrochemical oxidation of cysteine

Yeonwoo Kim$^1$, Sena Yang$^2$, Yeji Kang$^3$, Byung-Kwon Kim$^3$ & Hangil Lee$^3$

Transition metal-doped SnO$_2$ nanoparticles (TM-SnO$_2$) were synthesized by applying a thermosynthesis method, which first involved doping SnO$_2$ with Sb and then with transition metals (TM = Cr, Mn, Fe, or Co) of various concentrations to enhance a catalytic effect of SnO$_2$. The doped particles were then analyzed by using various surface analysis techniques such as transmission electron microscopy (TEM), X-ray diffraction (XRD), scanning transmission X-ray microscopy (STXM), and high-resolution photoemission spectroscopy (HRPES). We evaluated the catalytic effects of these doped particles on the oxidation of L-cysteine (Cys) in aqueous solution by taking electrochemical measurements and on the photocatalytic oxidation of Cys by using HRPES under UV illumination. Through the spectral analysis, we found that the Cr- and Mn-doped SnO$_2$ nanoparticles exhibit enhanced catalytic activities, which according to the various surface analyses were due to the effects of the sizes of the particles and electronegativity differences between the dopant metal and SnO$_2$.

SnO$_2$ is well known as one of the best smart materials and is used in an impressive range of applications including solar cells$^{1-4}$, catalyst supports$^{5-8}$, and chemical sensors$^{9-12}$. It has been particularly well regarded as a highly efficient catalyst in many gas conversions such as those involving CO, NO$_2$, and some hydrocarbons$^{13,14}$. Mechanistic studies of catalytic reactions using SnO$_2$ have been researched in several ways. It is, however, still necessary to enhance the catalytic effects of SnO$_2$, and attempts at such enhancements have been carried by inserting metals or anions, but such projects need to be mindful of the costs of the fabricated catalyst as well as its efficiency$^{15-17}$. Doping SnO$_2$ with Sb (to produce Sb@SnO$_2$) is one of the best-known modifications of the SnO$_2$ catalyst, and here the Sn 5s state was shown to contribute to the electrical conductivity because this state is related to the catalytic activity$^{18-20}$.

We have been pursuing a strategy of additional doping Sb@SnO$_2$ nanoparticles with transition metals to further enhance the catalytic activity of these nanoparticles. The procedure described below has in many cases been shown to increase their catalytic performance significantly. Transition metals such as Cr and Fe are the most feasible dopant candidates because they possess various oxidation states and thus contribute to the enhancement of the catalytic activities. For this purpose, in the current study, we inserted various transition metal ions (TM$^{+}$) into Sb@SnO$_2$ nanoparticles, and then systematically compared the catalytic activities of these metal-doped Sb@SnO$_2$ nanoparticles (TM-SnO$_2$).

In this work, the effects of doped transition metals, in particular Cr, Mn, Fe, and Co, on catalytic properties of the Sb@SnO$_2$ nanoparticles were studied when these metals were partially substituted into the nanoparticles. A one-pot synthesis was carried out to form uniformly sized (5–9 nm) catalyst nanoparticles and to maximize the 1 monolayer of the electron depletion layer. In particular, by commencing with TM-SnO$_2$ nanoparticles, we successfully fabricated them through a thermos-synthesis method and then assessed their catalytic capacities by oxidizing L-cysteine (Cys) under ultra-high vacuum (UHV) conditions with 365-nm-wavelength UV light illumination by using high-resolution photoemission spectroscopy (HRPES), and in the solution phase by using

---

$^1$Molecular-Level Interfaces Research Center, Department of Chemistry, KAIST, Daejeon, 34141, Republic of Korea. $^2$Center for Nano Characterization, Korea Research Institute of Standards and Science, Daejeon, 305-400, Republic of Korea. $^3$Department of Chemistry, Sookmyung Women’s University, Seoul, 04310, Republic of Korea. Correspondence and requests for materials should be addressed to B.-K.K. (email: kimbk@sookmyung.ac.kr) or H.L. (email: easyscan@sookmyung.ac.kr)
Results and Discussion

Synthesis. Sb@SnO₂ nanoparticles were prepared by the thermal synthesis method as described below. Sn: Sb ratio was selected as 100:5 to maximize the electrical conductivity which helps not only applying the electrical conductivity in SEM, HRPES measurements but also increasing the catalytic performance by shallowing the conduction band minimum. Around 5 mol% of Sb into the SnO₂ matrix, Sb@SnO₂ shows the lowest resistivity by n-type doping which was accomplished by the major contribution of Sn⁺¹ state below 22 mol% of Sb atom. XPS data in previous study shows no Sb³⁺ state at 3.8 mol% which means the hole transport effect can be excluded in those ratio. This structural design can be achieved the understanding of the only catalytic effects by electrons by fixing the major carrier component. That range of mole percent can also achieve UV absorption to confirm the photocatalytic activity. When Sn concentration is increased above 2 mol%, band gap of SnO₂ is reduced from 3.95 eV to below 3.65 eV which can actively absorb the UV wavelength.

TEM and XRD measurements. TEM images (Fig. 1) of 5 mole % TM-SnO₂ nanoparticles were obtained in order to determine the size distributions and shapes of the particles. These images revealed these particles to have very fine structures with dimensions mostly between 5 and 9 nm. All of the diffraction peaks of their XRD spectra were assigned to SnO₂ (cassiterite) since they were found to be consistent with the peaks at 2θ = 26.8°, 34.1°, 38.2°, 52.0°, 62.2°, 65.2°, and 71.6° corresponding to the (110), (101), (200), (211), (310), (112), and (202) reflections (JCPDS card No. 41–1455). The TEM images and XRD patterns showed the formation of single-phase SnO₂, indicative of successful doping of the transition metal into the SnO₂ nanoparticles without any phase separation and segregation. Sharp 2θ = 55.88 is correspond to the Si substrate which was mentioned in method section.

STXM measurements. Figure 2 shows the X-ray absorption spectra (XAS) and the corresponding stacked images (black regions of each inserted image) of the TM-SnO₂. Three small features between 490 and 507 eV and two small features between 498 and 502 eV were observed in the spectra at the Sn M₄,5₄-edges, which indicated the transition from the Sn⁺³ 3d state to the unoccupied p state. The features between 490 and 497 eV, and those between 498 and 502 eV, derived from Sn M₅ (3d₅/₂ and 3d₇/₂), respectively. The Sn M₄-edge measurements of all four TM-SnO₂ were very similar.

On the other hand, the O K-edge spectra region showed a quite different tendency. As shown in O K-edge region, peaks were observed at 533 eV and 536 to 540 eV, and were due to the transition from the O 1s state to the unoccupied p state, and from the O 2p state to the O 2p – Sn 5p hybrid orbital state, respectively. The shapes and intensities of the O K-edge peaks for Cr-SnO₂ were very similar to those for Mn-SnO₂ and to those for undoped SnO₂. However, the O K-edges of Fe-SnO₂ and Co-SnO₂ indicated more of the hybrid orbital (536 to 540 eV) than of the bare O 2p transition (533 eV), which may have been due to the Fe and Co dopants. The orbitals of these dopants each hybridized with the O 2p orbital according to the spectra. The oxidation states of the transition metal dopants were determined by the metal L-edges. For Cr-SnO₂, peaks were observed at 576.0 eV and 577.0 eV with a shoulder at 578.4 eV, corresponding to the Cr⁺³ L₃-edge. For Mn-SnO₂, a sharp peak was observed at 639.2 eV with a small feature at 640.7 eV, corresponding to the Mn⁺² L₃-edge. For Fe-SnO₂, a sharp peak was observed at 708.5 eV with a small peak at 706.6 eV, corresponding to the Fe⁺³ L₃-edge. For Co-SnO₂, a doublet was observed at 776.8 eV and 777.6 eV and was clearly separated from another doublet at 791.2 eV and 792.0 eV, and these peaks were confirmed as Co⁺³ L₃ and L₂-edges, respectively. These results indicated that STXM measurements can distinguish Cr-SnO₂ and Mn-SnO₂ from Fe-SnO₂ and Co-SnO₂.

Electrochemical redox reaction in the aqueous phase. CVs were obtained in a PBS solution containing 10 mM Cys at various types of GCEs irradiated by 365-nm-wavelength UV light. As shown in Fig. S1, a sluggish oxidation current was observed at a bare GCE because of the intrinsically slow electrochemical oxidation of Cys. To increase the current associated with the oxidation of Cys, GCEs modified with the TM-SnO₂-Nafion catalysts were fabricated and tested, with the results shown in Fig. 3. The currents associated with the oxidation of Cys were 14.2 (±1.7) and 10.5 (±1.6) μA when using the GCEs modified with the Cr-SnO₂ and Mn-SnO₂, respectively — significantly greater (i.e., 7.1 and 5.2 times greater) than the 2.0 μA value observed when using only the bare GCE (Fig. 3e). In contrast, the currents generated when using the Fe-SnO₂ and Co-SnO₂ were only 3.4 (±1.1), and 3.1 (±0.6) μA, respectively, which were slightly (1.7 and 1.5 times) but not significantly greater.
than that for the bare GCE. These results revealed the importance of the type of metal doped into the SnO$_2$ nanoparticles for catalyzing oxidation reactions, even when using small amounts (5%) of the doped metal, and specifically indicated the Cr-SnO$_2$ and Mn-SnO$_2$ to be good catalysts for the oxidation of Cys. Further investigations involving the optimization of the conditions are needed to selectively and sensitively detect Cys using Cr-SnO$_2$ and Mn-SnO$_2$ nanoparticles.

**Photocatalytic oxidation of L-cysteine.** We determined the catalytic activities of the TM-SnO$_2$ for the oxidation of Cys molecules. The surface-sensitive S 2p core-level HRPES spectra were acquired from the products of the exposure of a 180 L of Cys to the actual amount of oxygen used and 365-nm-wavelength UV light in the presence of each type of TM-SnO$_2$ (Fig. 4(a–d)). As shown in these figures, three distinct 2p$_{3/2}$ peaks were observed, at 161.5, 162.9, and 168.6 eV, which corresponded to the thiol group (-SH; denoted as S1), the bound state (denoted as S2), and sulfonic acid (SO$_3$H) (denoted as S3), respectively. Since sulfonic acid has been shown to be an oxidation product of the thiol group$^{37,38}$, we monitored the oxidation of Cys by measuring the ratio of the intensity of S1 to that of S3 for each of the four types of TM-SnO$_2$. The results of these measurements (Fig. 4(a–d)) confirmed Cr-SnO$_2$ and Mn-SnO$_2$ to be effective catalysts. To determine the effects of metal doping on the catalytic performance, the ratio of S3 to S1 was determined for 5% mole dopant concentrations (Fig. 4(e)). In particular, Cr-SnO$_2$ and Mn-SnO$_2$ exhibited clear enhancements in photocatalytic activity, and these results were closely correlated with the EC results. We also performed experiments involving the conversion of CO to CO$_2$ (Fig. S2) as described in Supplementary Information text to clarify the trend of the catalytic activities for these TM-SnO$_2$ at various TM doping concentrations.

Through the characterizations of the electronic structures and of catalytic oxidation reactions for the four TM-SnO$_2$, we found that two quite different catalytic measurements — the rate of electrochemical oxidation of Cys in aqueous solution and the rate of catalytic oxidation in ultra-high vacuum conditions — showed the same trends. In both cases, the Cr-SnO$_2$ and Mn-SnO$_2$ showed the highest and second highest catalytic activities, respectively. These same trends were found despite the measurements conditions (in aqueous and under UHV
Figure 2. XAS spectra of (top panels) Sn $M_{45}$-edges, (middle panels) O $K$-edge, and (bottom panels) doped transition metal $L_{2,3}$-edge, and (middle panels) the corresponding stacked images, for 5 mole% Cr-SnO$_2$, Mn-SnO$_2$, Fe-SnO$_2$, and Co-SnO$_2$ nanoparticles.

Figure 3. (a–d) CVs (at a scan rate of 50 mV/s) in PBS containing 10 mM Cys at a bare GCE (black lines) or GCEs modified (red lines) with 5 mole% (a) Cr-SnO$_2$, (b) Mn-SnO$_2$, (c) Fe-SnO$_2$, and (d) Co-SnO$_2$ nanoparticles. (e) Catalytic currents resulting from the electrochemical oxidation of Cys for the various types of TM-SnO$_2$ nanoparticles.
condition) being so different. These results suggest the effect of the identity of the doped metal on the catalytic activity of SnO$_2$ to be independent of the environmental conditions.

Previous results about TM-SnO$_2$ reported SnO$_2$ nanoparticles doped with only one type of metal and reported only a single measurement phase (in aqueous or under UHV condition). Therefore, usual concepts were used to explain observed increases in catalytic activity upon metal doping. For example, for catalytic oxidation experiments carried out in vacuum conditions, band gap theory was frequently used to explain the catalytic activities of the doped TM-SnO$_2$ due to the bandgap narrowing from the 3.6 eV value for the bare SnO$_2$ nanoparticles upon being doped$^{39,40}$. In general, the bandgap narrowing of a TM-SnO$_2$ leads to enhanced photocatalytic activity. However, it showed slightly enhanced catalytic activities upon being doped with a metal. This explanation for the increase in the catalytic activity is well established in the field of photocatalysis. In contrast, increases in the electrochemical catalytic activities in the aqueous phase resulting from doping a catalyst with a metal have been explained by the doping causing an increase in the conductivity. Doping Fe into SnO$_2$, for example, has been shown to increase its conductivity, and this increase in conductivity has been offered as an explanation for the observed increase in catalytic activity$^{41}$. When analyzing the results of each individual experimental condition (aqueous condition or under vacuum) on its own, each corresponding explanation (narrowing the bandgap or increasing the conductivity) appears persuasive. However, in our experiments, the results showed the same trends for both conditions, and hence another explanation is needed for why doping causes an increase in catalytic activity.

One possible alternative explanation is the difference of the electronegativity between the doped transition metals and SnO$_2$. The electronegativities (Allen scale) of Cr, Mn, Fe, and Co are 1.56, 1.60, 1.64, and 1.70 respectively$^{42,43}$, while those of Sn and O are 1.72 and 3.52, respectively. When these transition metals, which have lower electronegativity values than Sn, are doped into the SnO$_2$, hot spots for the oxidation of Cys are generated. Cys has electron-rich thiol (-SH) and amine (-NH$_2$) groups that bind more strongly to these hot spots, which are positively charged, than to adjacent atoms. These hot spots are fairly attractive for the oxidation of the Cys.

Another reasonable explanation is that according to the O K-edge XAS region shown in Fig. 2, a higher proportion of less hybridized oxygen states (533 eV) appeared in Cr-SnO$_2$ and Mn-SnO$_2$ than in the other TM-SnO$_2$. Those transition of the doped metal 3d to the O 2p unoccupied state can facilitate the removal of oxygen atoms from the SnO$_2$ nanoparticles and promote the catalytic oxidation of Cys because oxygen vacancy site of SnO$_2$ is an active site. According to the HRPES data of TM-SnO$_2$ by increment of the amount of the dopants (Figs S3–S6), only Cr-SnO$_2$ (Fig. S3) and Co-SnO$_2$ (Fig. S6) show the additional state in valence band region. However, there have no relationship between the new states and catalytic performance which means the reaction did not occur by the hole transfer. The reasonable interpretation of binding sites and reactive charge is that the hot electrons are transferred to the oxygen vacancy sites (electron trap sites) to enhance the photo- and electro-chemical activities. In Figs S3–S6, there have 531.9 eV in O 1s spectra which are assigned as oxygen vacancy sites$^{44}$. The starting point of those materials are SnO$_2$ with 5 mol% of Sb doping. This structural design has the electron-enhanced structure by the 100% Sb$^{5+}$ doping (n-type SnO$_2$). In addition to this structural design, electronegativity model supports the high catalytic performance by introduction of additional transition metals. In the cases of Fe-SnO$_2$ and Co-SnO$_2$, the electronegativities of Fe and Co are only slightly lower than that of Sn, and the proportion of O 2p orbitals that are hybridized with Sn 5d orbitals is larger, and tightly bound to the SnO$_2$ nanoparticle. Correspondingly, doping the SnO$_2$ nanoparticle with either Fe or Co yielded a smaller increase in the catalytic activities for Cys oxidation than did doping with Cr or Mn. Hence, we believe that this electronegativity hypothesis and less hybridization

Figure 4. HRPES S 2p core-level spectra of the products of the photocatalytic oxidations of Cys (a 180 L solution) carried out in the presence of 5 mole% (a) Cr-SnO$_2$, (b) Mn-SnO$_2$, (c) Fe-SnO$_2$, and (d) Co-SnO$_2$ nanoparticles. (e) Values of the S3 to S1 ratio (see text), the four types of TM-SnO$_2$ nanoparticles, resulting from 180 L exposure of Cys solutions to 365-nm-wavelength UV light, in order to assess the photocatalytic activity of each type of nanoparticle towards the oxidation of cysteine.
result between the doped metal 3d to the O 2p are fairly straightforward and reasonable, but requires further experiments to be proven.

Conclusion
The results of catalytic oxidation through the EC measurements and the HRPES analysis, which showed the catalytic activities of Cr-SnO2 and Mn-SnO2 to be superior to those of the other metal-doped nanoparticles or bare SnO2 nanoparticles. It is in good agreement with our STXM measurements indicated the dopant amounts of Cr-SnO2 and Mn-SnO2 to be better arranged than those of SnO2, Fe-SnO2, and Co-SnO2. As a result, Cr-SnO2 and Mn-SnO2 showed higher catalytic activities due to the electronegativity differences and less hybridizations between the dopant metals and SnO2, which are induced by increasing the oxygen vacancy site of SnO2.

Methods
Preparation of precursor solutions. We prepared the precursor solutions by carrying out one-pot syntheses. 10 mmol of tin (IV) chloride pentahydrate (SnCl4∙5H2O, 98%) and 0.5 mmol of antimony (III) chloride (SbCl3, >99.0%) were added to 10 ml of 2-methoxy ethanol (≥99.9%). Dopants (M) were added in the form of the transition metal nitrate n-hydrate (TM(NO3)2∙nH2O) in desired amounts that were determined by calculating the mole fraction of the metal dopant according to the equation (moles of TM)/(moles of Sn). Specifically, Cr(NO3)3∙9H2O (99%), Mn(NO3)2∙xH2O (98%), Fe(NO3)3∙9H2O (98%), and Co(NO3)2∙6H2O (≥98%) were used as dopants. All substances were purchased from Sigma Aldrich. Precursor solutions were stirred for 10 minutes. L-cysteine (Sigma Aldrich, 97% purity), and Nafion (Sigma Aldrich, 5 wt% in a low-molecular-weight aliphatic alcohol and water) were purchased from Sigma-Aldrich. Phosphate-buffered saline (PBS) tablets were purchased from Gibco.

Preparation of the TM-SnO2 films. Silicon wafers (10 mm × 10 mm) were washed with ethanol, acetone, and distilled water for several times. Then, they were sonicated and dried by N2 air followed by oxygen plasma treatment for 3 minutes. Each TM-SnO2 precursor solution was then spin coated onto respective silicon wafers at 2000 rpm for 30 seconds. Subsequently, the spin-coated films were annealed at 800°C for 5 hours (heating rate of 5°C/min) in ambient pressure.

Preparation of the TM-SnO2. Nanoparticles were prepared by annealing each of the precursor solutions directly. Precursor solutions were put into Al2O3 crucibles and then annealed at 800°C for 5 hours using the same annealing procedure as used for the film synthesis.

Fabrication of TM-SnO2-Nafion-modified GCE and electrochemical measurements of Cys oxidation. The electrochemical oxidation of Cys was investigated using glassy carbon electrodes (GCEs) modified with TM-SnO2 nanoparticles. For each M, a mass of 4.0 mg of TM-SnO2 nanoparticles was dispersed into 1.0 ml of distilled water containing 50 μl Nafion, and then mixed by using an ultrasonic processor (Wise cleaner, DAIHAN Sci., Wonju, Korea) for 10 minutes to obtain the homogeneous TM-SnO2-Nafion mixture. After that, a volume of 20 μl of the mixture was placed on a GCE, and was dried at 75°C in a pre-heated oven for 30 minutes. A cyclic voltammogram (CV) of 10 mM Cys in PBS was obtained for each TM-SnO2-Nafion modified GCE.

Characterizations. The morphologies of the samples were characterized by performing TEM (FEI Tecnai G2 F30 S-Twin) at an acceleration voltage of 300 kV. XRD patterns of the TM-SnO2 nanoparticles were obtained using Ni-filtered Cu-Kα radiation from a Rigaku D/MAX-A diffractometer. STXM was performed at the 10 A beamline at the Pohang Accelerator Laboratory (PAL). A Fresnel zone plate with an outermost zone width of 25 nm was used to focus the X-rays onto the TM-SnO2 nanoparticles on the TEM grids. Image stacks were acquired using X-ray absorption spectroscopy (XAS) to extract the transition metal L-edge, Sn M-edge, and O K-edge spectra. HRPES experiments were performed at the 8A1 beamline at PAL with an electron analyzer (Physical Electronics, PHI-3057). The binding energies of the core level spectra were determined with respect to the binding energy (Eg = 840 eV) of the clean Au 4f core level for the same photon energy. The electrochemical experiments were performed by using a CHI617B potentiostat (CH Instruments, Austin, TX) with a three-electrode cell placed in a Faraday cage. A GCE with a diameter of 2 mm was used as the working electrode, a Pt wire with a diameter of 0.5 mm was used as the counter electrode, and the reference electrode was Ag/AgCl (3 M KCl). All working electrodes were polished with alumina (0.05 μm) paste on micro-cloth pads (Buehler, Lake Bluff, IL) prior to use. After measurements of electrochemical reaction of Cys with TM-SnO2-Nafion modified GCE, the catalytic activities were compared. The current values at 0.7 V (because of new small shoulder current) were selected to compare the catalytic activities of TM-SnO2 nanoparticles.

References
1. Ke, W. et al. Low-Temperature Solution-Processed Tin Oxide as an Alternative Electron Transporting Layer for Efficient Perovskite Solar Cells. J. Am. Chem. Soc. 137, 6730–6733 (2015).
2. Wang, H. & Rogach, A. L. Hierarchical SnO2 Nanostructures: Recent Advances in Design, Synthesis, and Applications. Chem. Mater. 26, 123–133 (2014).
3. Snaith, H. J. & Ducati, C. SnO2-Based Dye-Sensitized Hybrid Solar Cells Exhibiting Near Unity Absorbed Photon-to-Electron Conversion Efficiency. Nano Lett. 10, 1259–1265 (2010).
4. Ito, S., Makari, Y., Kitamura, T., Wada, Y. & Yanagida, S. Fabrication and characterization of mesoporous SnO2/ZnO-composite electrodes for efficient dye solar cells. J. Mater. Chem. 14, 385 (2004).
5. Xu, J., Liu, G., Li, J. & Wang, X. The electrocatalytic properties of an IrO2/SnO2 catalyst using SnO2 as a support and an assisting reagent for the oxygen evolution reaction. Electrochim. Acta 59, 105–112 (2012).
6. Yu, K., Wu, Z., Zhao, Q., Li, B. & Xie, Y. High-Temperature-Stable Au@SnO2 Core/Shell Supported Catalyst for CO Oxidation. J. Phys. Chem. C 112, 2244–2247 (2008).
8. Oh, H.-S., Nong, H. N. & Strasser, P. Preparation of Mesoporous Sb-, F-, and In-Doped SnO₂ Bulk Powder with High Surface Area for Use as Catalyst Supports in Electrolytic Cells. *Adv. Funct. Mater.* 25, 1074–1081 (2015).

9. Göpel, W. & Schierbaum, K. D. SnO₂ sensors: current status and future prospects. *Sensors Actuators B Chem.* 26, 1–12 (1995).

10. Kim, H.-R. et al. The Role of NiO Doping in Reducing the Impact of Humidity on the Performance of SnO₂-Based Gas Sensors: Synthesis Strategies, and Phenomenological and Spectroscopic Studies. *Adv. Funct. Mater.* 21, 4456–4463 (2011).

11. Mei, L., Chen, Y. & Ma, J. Gas Sensing of SnO₂ Nanocrystals Revisited: Developing Ultra-Sensitive Sensors for Detecting the H₂S Leakage of Biogas. *Sci. Rep.* 4, 6028 (2014).

12. Das, S. & Jayaraman, V. SnO₂: A comprehensive review on structures and gas sensors. *Prog. Mater. Sci.* 66, 112–355 (2014).

13. Yoo, J. & Wachsmann, E. D. NO₂ response of Cr₂O₃- and SnO₂-based potentiometric sensors and temperature-programmed reduction evaluation of the sensor elements. *Sensors Actuators B Chem.* 123, 915–921 (2007).

14. Maeng, S. et al. SnO₂ Nanoslab as NO₂ Sensor: Identification of the NO₂ Sensing Mechanism on a SnO₂ Surface. *ACS Appl. Mater. Interfaces* 6, 357–363 (2014).

15. Feng, Y. et al. The Magnetic and Optical Properties of 3d transition metal doped SnO₂ Nanosheets. *RSC Adv.* 5, 23406–23412 (2015).

16. Ji, L., Cai, W. & Wang, H. Metal ion-doped SnO₂ ordered porous films and their strong gas sensing selectivity. *Appl. Phys. Lett.* 96, 103115 (2010).

17. Iwakura, C., Inai, M., Uemura, T. & Tamura, H. The anodic evolution of oxygen and chlorine on foreign metal-doped SnO₂ film electrodes. *Electrochim. Acta* 26, 579–584 (1981).

18. Mishra, K. C., Johnson, K. H. & Schmidt, P. C. Electronic structure of antimony-doped tin oxide. *Phys. Rev. B* 51, 13972–13976 (1995).

19. Batzill, M. & Diebold, U. The surface and materials science of tin oxide. *Prog. Surf. Sci.* 79, 47–154 (2004).

20. Kida, T., Fujiyama, S., Suematsu, K., Yuasa, M. & Shimanoe, K. Pore and Particle Size Control of Gas Sensing Films Using SnO₂ Nanoparticles Synthesized by Seed-Mediated Growth: Design of Highly Sensitive Gas Sensors. *J. Phys. Chem. C* 117, 17574–17582 (2013).

21. Punnoose, A. & Hays, J. Possible metamagnetic origin of ferromagnetism in transition-metal-doped SnO₂. *J. Appl. Phys.* 97, 10D321 (2005).

22. Zhang, G. & Liu, M. Effect of particle size and dopant on properties of SnO₂-based gas sensors. *Sensors Actuators B Chem.* 69, 144–152 (2000).

23. Chen, X., Ko, S.-K., Kim, M. J., Shin, I. & Yoon, J. A thiol-specific fluorescent probe and its application for bioimaging. *Chem. Commun.* 46, 2751 (2010).

24. Wu, S. et al. Selective electrochemical detection of cysteine in complex serum by graphene nanoribbon. *Biosens. Bioelectron.* 32, 293–296 (2012).

25. Wang, W. et al. Detection of Homocysteine and Cysteine. *J. Am. Chem. Soc.* 127, 15949–15958 (2005).

26. Hager, G. & Brolo, A. G. Adsorption/desorption behaviour of cysteine and cystine in neutral and basic media: electrochemical evidence for differing thiol and disulfide adsorption to a Au(111) single crystal electrode. *J. Electroanal. Chem.* 550–551, 291–301 (2003).

27. Yang, L. et al. Efficient hydrogen evolution over Sn doped SnO₂ photocatalyst sensitized by Eosin Y under visible light irradiation. *Nano Energy* 36, 331–340 (2017).

28. Terrier, C., Chatelon, J., Berjoan, R. & Roger, J. Sn-doped SnO₂ transparent conducting oxide from the sol-gel dip-coating technique. *Thin Solid Films* 263, 37–41 (1995).

29. Terrier, C., Chatelon, J. P. & Roger, J. A. Electrical and optical properties of Sn-doped SnO₂ thin films obtained by the sol-gel method. *Thin Solid Films* 295, 95–100 (1997).

30. Wang, X. et al. Oxidation and detection of L-cysteine using a modified Au/Nafion/glass carbon electrode. *Sci. China Chem.* 54, 521–525 (2011).

31. Tan, W. T., Bond, A. M., Ngooi, S. W., Lim, E. B. & Goh, J. K. Electrochemical oxidation of L-cysteine mediated by a fullerene-C60-modified carbon electrode. *Anal. Chem.* 491, 181–191 (2003).

32. Shahrokshian, S. & Bozorgzadeh, S. Electrochemical oxidation of dopamine in the presence of sulphhydril compounds: Application to the square-wave voltammetric detection of penicillamine and cysteine. *Electrochim. Acta* 51, 4271–4276 (2006).

33. Chuvchenkova, O. A. et al. Synchrotron studies of SnO₂ wire-like crystals. *J. Surf. Invest. X-ray, Synchrotron Neutron Tech.* 8, 111–116 (2014).

34. Baumann, T. F., Kucheyes, S. O., Gash, A. E. & Satcher, J. H. Facile Synthesis of a Crystalline, High-Surface-Area SnO₂ Aerogel. *Adv. Mater.* 17, 1546–1548 (2005).

35. Moreno, M. S., Egerton, R. F. & Midgley, P. A. Differentiation of tin oxides using electron energy-loss spectroscopy. *Phys. Rev. B* 69, 233304 (2004).

36. Wang, D. et al. Layer by layer assembly of sandwiched graphene/SnO₂ nanorod/carbon nanostructures with ultrahigh lithium ion storage properties. *Energy Environ. Sci.* 6, 2900 (2013).

37. Meyers, D. et al. Zhang-Rice physics and anomalous copper states in A-site ordered perovskites. *Sci. Rep.* 3, 1834 (2013).

38. Qiao, R., Chin, T., Harris, S. J., Yang, S. & Yang, W. Spectroscopic fingerprints of valence and spin states in manganese oxides and fluorides. *Curr. Appl. Phys.* 13, 544–548 (2013).

39. Liu, X. et al. Distinct charge dynamics in battery electrodes revealed by in situ and operando soft X-ray spectroscopy. *Nat. Commun.* 4, 1–8 (2013).

40. Rodella, C. B. et al. Physical and chemical studies of tungsten carbide catalysts: effects of Ni promotion and sulphonated carbon. *RSC Adv.* 5, 23874–23885 (2015).

41. Suganuma, S. et al. Hydrolysis of Cellulosic Amorphous Carbon Bearing SO 3 H, COOH, and OH Groups. *J. Am. Chem. Soc.* 130, 12787–12793 (2008).

42. Salah, N., Habib, S., Azam, A., Ansari, M. S. & Al-Shawafi, W. M. Formation of Mn-doped SnO₂ Nanoparticles Via the Microwave Technique: Structural, Optical and Electrical Properties. *Nanomater. Nanotechnol.* 1 https://doi.org/10.5772/62520 (2013).

43. Subramanym, K., Sreelekha, N., Murali, G., Reddy, D. A. & Vijayalakshmi, R. P. Structural, optical and magnetic properties of Cr doped SnO₂ nanoparticles stabilized with polyethylene glycol. *Phys. B Condens. Matter* 454, 86–92 (2014).

44. Shi, S., Gao, D., Xu, Q., Yang, Z. & Xue, D. Singly-charged oxygen vacancy-induced ferromagnetism in mechanically milled SnO₂ powders. *RSC Adv.* 4, 45467–45472 (2014).

**Acknowledgements**

This research was supported by the National Research Foundation of Korea (NRF) funded by the Korea government (MSIP) (No. 2017R1A2A2A05001140). Additionally, this research is financially supported by the Ministry of Trade, Industry and Energy (MOTIE) and Korea Institute for Advancement of Technology (KIAT) through the International Cooperative R&D program (N05310009, “Horizon2020 Kor-EU collaborative R&B on AECenano Toolbox”) as part of the European Commission Horizon 2020 Programme under grant agreement NMBP-26-2016-720952.
Author Contributions
H.L. and B.K. designed the experiments and wrote the manuscript. Y.K., S.Y. and Y.K. performed the experiments.

Additional Information
Supplementary information accompanies this paper at https://doi.org/10.1038/s41598-018-30962-0.

Competing Interests: The authors declare no competing interests.

Publisher’s note: Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article’s Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article’s Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2018