Sensitivity Enhancement of Transition Metal Dichalcogenides/Silicon Nanostructure-based Surface Plasmon Resonance Biosensor

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In this work, we designed a sensitivity-enhanced surface plasmon resonance biosensor structure based on silicon nanosheet and two-dimensional transition metal dichalcogenides. This configuration contains six components: SF10 triangular prism, gold thin film, silicon nanosheet, two-dimensional MoS2/MoSe2/WS2/WSe2 (defined as MX2) layers, biomolecular analyte layer and sensing medium. The minimum reflectivity, sensitivity as well as the Full Width at Half Maximum of SPR curve are systematically examined by using Fresnel equations and the transfer matrix method in the visible and near infrared wavelength range (600 nm to 1024 nm). The variation of the minimum reflectivity and the change in resonance angle as the function of the number of MX2 layers are presented respectively. The results show that silicon nanosheet and MX2 layers can be served as effective light absorption medium. Under resonance conditions, the electrons in these additional dielectric layers can be transferred to the surface of gold thin film. All silicon-MX2 enhanced sensing models show much better performance than that of the conventional sensing scheme where pure Au thin film is used, the highest sensitivity can be achieved by employing 600 nm excitation light wavelength with 35 nm gold thin film and 7 nm thickness silicon nanosheet coated with monolayer WS2.

Surface plasmon resonance (SPR)-based biosensors have attracted great attention as one of the leading optical sensing technologies during the last two decades due to their unique advantages such as real-time and label-free detection1–3. They played an important role in monitoring various biomolecular interactions like protein bindings and DNA hybridization4,5. Further applications such as pharmacology and early disease diagnostics are promising if the SPR sensitivities have a drastic improvement6. The first observation of the SPR dated back to 1902, however, the complete explanations on this phenomenon were not provided by the scientific community until 1968 when Kretschmann, Raether and Otto proposed effective excitation configurations and abundant theoretical demonstration7,8. The surface plasmons (SPs) can be considered as electron density waves that propagate at the interface between metal and dielectric. In order to excite the surface plasmon waves (SPW), the incident photons or electrons should oscillate with the free electrons on the metal surface to form the resonances. Only p-polarized light (TM waves) contributes to the excitation of the SPWs, while s-polarized light (TE waves) could act as the reference signals. When the horizontal component of the incident wave vector kx (i.e., the evanescent wave vector)

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matches with the surface plasmon wave vector $k_{sp}$, the surface plasmon resonance phenomenon occurs, as shown in equation (1),

$$k_s = k_0 \sin \theta_{sc} = k_{sp}$$  \hspace{1cm} (1)

where $k_0$ represents the incident wave vector in free space and $\theta_{sc}$ denotes the incident angle. Here, the incident angle is known as the resonance angle (or SPR angle). When the excitation light wavelength was fixed, one can obtain a SPR curve with a dip by scanning the incident angle and monitoring the reflectivity. The minimum of the reflectivity (nearly zero) corresponded to the reflected intensity at the resonance angle. When the SPW was excited, part of the incident optical energy was converted into the SPW resonance energy, leading to the drastic decrease in the reflected intensity. Ideally, all incident energy should be absorbed to support the resonant oscillations and result in a strong evanescent field. For the SPR biosensor, the SPR angle serves as an important output signal for the angular interrogation system. Since the refractive index changes of the sensing layer that are induced by the adsorption of biomolecules on the sensing surface would lead to a redistribution of SPR electromagnetic field, a significant SPR angle shift could be obtained and collected through an optical detector. This unique characteristic enables SPR biosensors to show excellent performances for real-time and label-free detections. However, the sensitivity is known to be limited especially when the weight of the biomolecules analyte is less than 500 Da9. Various methods are provided to enhance the sensitivity: Silver thin film as SPR sensing substrate was demonstrated to have better performance than that with gold in sensitivity, however the weak chemical stability of silver impedes its further development10; In addition, the coating of an additional dielectric nanolayer on the sensing film was reported to exhibit sensitivity enhancement effects11.Attributed to the excellent optoelectronic properties and the advanced fabrication techniques (e.g. graphene growth on metallic substrates)10-11, graphene layers have been employed to enhance the SPR biosensor sensitivity. As reported in Wu et al.12 study, 10 layers of the graphene coated on the gold sensing surface can improve the sensitivity by 25%. The enhanced effect of single nanomaterial toppings, however, still seems insufficient for further development of SPR biosensor applications. Thus, hybrid nanostructures such as silicon-graphene and MoS2-graphene thin film were investigated and revealed prominent sensitivity enhanced effect13-15. The silicon nanosheet is able to enhance the sensitivity of the SPR biosensor because of its large real value of the refractive index. It also serves as a protective layer of metal film to improve the overall system stability16,17. Recently, the emerging two-dimensional (2D) transition metal dichalcogenides (TMDCs) have been widely used in transistors and photodetectors due to the remarkable electrical and optical properties. The TMDCs family consists of more than forty compounds that generally defined as MX2, where M stands for the transition metal from group IV to group VII, like Nb, Ta, Mo and W and the X denotes the chalcogen such as S, Se and Te. Monolayer MX2 contains three atomic layers where the transition metal layer is sandwiched by two chalcogens layers. Each layer is stacked via van der Waals forces. In this work, we focus on the group-IV semiconductor dichalcogenides MoX2 and WX2, namely Molybdenum disulfide (MoS2), Molybdenum diselenide (MoSe2), Tungsten disulfide (WS2) and Tungsten diselenide (WSe2). The rapid fabrication development of the high quality (i.e., large areas, highly uniform) individual 2D MX2 layers by chemical exfoliation method promotes versatility of 2D MX2 in various fields, such as photonics, electronics, energy storage, catalysis and even biomedical applications18-20. Although the properties of bulk MX2 have been investigated for decades, the successful translation of 2D MX2 in optoelectronics and nanoelectronics is still remained in a stagnant stage. It is well known that when the bulk material is downscaled to a single layer, the bandgap transition would be gradually shifted from indirect to direct state20,21. This can be explained by the quantum confinement and resulted from the change in hybridization between orbital of Mo/W and X atoms20,22. The electronic bands of these monolayer materials are comparable to that of silicon (1.1 eV), which allows good performance in digital transistors23. Furthermore, these characteristics also affect the photophysical properties. For the semiconductor materials with a direct bandgap like MoX2 and WX2, the photons can be directly absorbed or emitted if the external energy is larger than the bandgap. However, for the indirect bandgap materials, photons could not be absorbed directly, since additional phonons were needed to provide the energy for the electron to surpass the intermediate state and transfer the momentum to the crystal lattice. Therefore, the photon absorption process for direct bandgap materials are much more efficient than that of indirect bandgap materials. The single layered MoS2, which known as “beyond graphene” 2D nanocrystals material has attracted a great deal of attention. Due to the quantum confinement effects, the monolayer MoS2 has a direct bandgap of 1.8 eV, while bulk MoS2 has an indirect bandgap of 1.2 eV22,24. This allows 2D MoS2 to be used in nano-transistor channel with a large switching ratio25 and in photodetectors with a high responsivity up to $5 \times 10^4$ AW$^{-1}$.26 Monolayer MoS2 also plays a key role for enhancing the sensitivity of SPR optical sensor13,27. Basically, a MoS2 enhanced hybrid nanostructure SPR biosensor can drastically improve detection limit of the device by using phase modulation technique. Similar to MoS2, the confinement of charge carriers on the horizontal atomic plane can gradually enlarge the energy gap of the WS2, atomic layers27-29. Theoretical studies show that monolayer WS2 has better performance compared to MoS2 in terms of enhancing the carrier mobility when they serve as channels in the transistor. This is attributed to the lower electron effective mass of WS2 compared to other MX2 materials30. In the research field of photoelectronics, nano-scaled WS2 also showed outstanding performance. For example, 2D heterostructures consisting of WS2, MoS2, GaSe and graphene can exhibit photovoltaic effects with external quantum efficiencies up to 30%27,31. Moreover, the employment of WS2 monolayer in plasmonic applications has enhanced the efficiency appreciably32. WSe2 nanolayers also attracted wide attention in photoelectronics, due to its fine absorption and emission features33 as well as the strong exciton charging effect34. Koperski et al.35 reported a comprehensive study of optical micro-spectroscopy based on thin layers of WSe2, where narrow emission lines ($\sim 100$ µeV linewidth) were obtained because the monolayer WSe2 can generate luminescence within the same energy range. Ross et al.36 demonstrated a monolayer WSe2 p–n junction based LED structure that produced effective injections of electrons and holes due to the high optical quality. It yielded bright electroluminescence
with much smaller injection current and linewidth compared with MoS2-based structures. The monolayer MoSe2 also provides promising optical applications because of its direct bandgap (identified at 1.55 eV)\(^3^7\). It is reported that the MoSe\(_2\) nanostructures show reversible and sensitive photo-responsive (PR) properties with the PR current values reaching up to \(2.55 \times 10^{-5}\) A\(^3^8\). The lateral heterojunctions within monolayer MoSe\(_2\)-WSe\(_2\) are visible under the optical microscope and show enhanced photoluminescence\(^3^9\). The strong photoluminescence emission is caused by the transition from an indirect band gap semiconductor of bulk material to a direct band gap semiconductor in atomically thin form\(^4^0\). These properties provided a solid foundation for atomic thin semiconductor dichalcogenides as promising candidates for next generation nanoelectronics, and optoelectronics\(^4^1\).

Based on these remarkable properties of silicon and group-IV semiconductor dichalcogenides, we propose a new configuration for sensitivity enhanced SPR biosensors based on silicon-MX\(_2\) heterostructures. As shown in Fig. 1, based on the Kretschmann attenuated total reflection (ATR) configurations, gold thin film is attached at the bottom of the SF10 prism followed by silicon nanosheet and 2D MX\(_2\). The 2D MX\(_2\) layers which are directly contacted with biomolecular analyte have dual effects: (i) as the signal-enhanced layer due to a high charge transfer efficiency from the MX\(_2\) surface to the Au surface\(^4^2\)–\(^4^4\); (ii) as sensing platform to capture the biomolecules through the van der Waals interaction\(^4^5\),\(^4^6\).

**Results and Discussion**

In the ATR configuration, according to the principle of energy conversation the sum of the absorption \(A\), reflectance \(R\) and transmittance \(T\) must be equal to 1 (i.e., \(A + R + T = 1\)) assuming no energy loss besides the materials absorption. Under the ATR condition, \(T\) is always equal to zero, hence the absorption of the system can be reduced to \(A = 1 - R\). When the SPs were excited, the reflectance \(R\) gradually decreased until it reached a minimum. The minimum \(R\) was close to zero indicating that the incident energy was almost completely absorbed by the layered materials. Therefore, maximum incident light energy transfer to the evanescent wave is required in order to achieve the best SPR enhancement performance\(^4^5\),\(^4^7\),\(^4^8\).

**Optimization of number of MX\(_2\) layers.** To optimize the number of MX\(_2\) layers, we plotted the resonance depths (i.e., the value of minimum reflectivity) change as a function of number of MX\(_2\) layers with various thickness of silicon nanosheet (i.e., 0 nm, 5 nm, 7 nm) and gold thin film (i.e., 30 nm, 35 nm, 40 nm, 50 nm). Figures 2a–c, 3a–c, 4a–c, 5a–c and 6a–c show the simulation results where different excitation wavelengths at 600 nm, 633 nm, 660 nm, 785 nm and 1024 nm were used. In general, two features for the SPR curves were observed: (i) When the silicon thickness was fixed and decreasing the gold thickness, the SPR dips redshifted with larger number of MX\(_2\) layers; similarly, when the thickness of gold thin film was fixed and decreasing the thickness of silicon, the reflectivity dips would also redshifted with the larger number of MX\(_2\) layers. This indicates that both silicon and MX\(_2\) layers can serve as absorption mediums and transfer energy to the gold thin film. (ii) It is worth noting that when the gold thickness was fixed at 50 nm, no MX\(_2\) is required to achieve the minimum reflectivity since 50 nm is the optimized thickness of gold thin film based on the conventional Kretschmann ATR configuration\(^4\). By analyzing the relationship between reflectivity and the number of MX\(_2\) layers, optimization of the number of MX\(_2\) layers was possible by selecting the specific number of layers that corresponded to the minimum resonance depths (i.e., the value of reflectivity that closest to zero). The detailed optimized number of MX\(_2\) layers with minimum reflectivity \(\text{min } R\) less than 0.03 were listed in Tables S1–S20 (Supplementary Information).

**Thickness Optimization of gold and silicon layers.** The optimized thickness of gold thin film and silicon nanosheet were obtained from the optimization of the sensitivity and FWHM. The SPR sensitivity serves as the key characteristic of SPR biosensor is defined in equation (13) (See the Methods section). When the refractive index change of the biomolecules analyte layer (\(\Delta n_{\text{bio}} = 0.005\)) was fixed, the sensitivity was governed by the change in resonance angle that acquired before and after the adsorption of biomolecules on the surface of the MX\(_2\) layers surface. Upon investigation of the variation of change in the resonance angle as a function of the number of MX\(_2\) layers as shown in Figs 2d–f to 6d–f three main features were observed: (i) As the number of MX\(_2\) layers...
increased, the change in resonance angle increased until it reached a maximum followed by a rapid decreasing to quasi-zero. (ii) When the excitation wavelength and gold thickness were fixed, the reduced thickness of silicon nanosheet caused the peak of the change in resonance angle to shift toward a larger number of MX_2 layers; similarly, if the excitation wavelength and the silicon thickness were fixed, the decreased thickness of gold thin film also led to a peak shift to a larger number of MX_2 layers. (iii) With fixed thickness of silicon and gold, the redshifted excitation wavelengths induced the peak shift of change in resonance angle towards the larger number of MX_2 layers. Similar behaviors were also observed in Figs 2a–c to 6a–c (i.e., at longer excitation wavelengths, a larger number of MX_2 was required to achieve the minimum reflectivity dip). This showed that the angular sensitivity decreases with longer excited wavelengths, thereby higher refractive index materials are required to enhance the evanescent field.

With optimized number of MX_2 layers, the angular sensitivity and FWHM of the four silicon-MX_2 enhanced models were then analyzed under five different excitation wavelengths. When the excitation wavelength was set at 600 nm, the highest sensitivity of 155.68 Deg/RIU was achieved with optimum parameters of 35 nm thickness gold film, 7 nm thickness silicon and monolayer WS_2. The FWHM was 17.4644 Deg with a minimum reflectivity of $2.5592 \times 10^{-2}$ as shown in Table S1 (Supplementary Information). However, with the same thickness parameters and excitation wavelength, the highest sensitivity of silicon-MoSe_2 enhanced model was only 104.56 Deg/RIU with a minimum reflectivity of $4.7627 \times 10^{-3}$ as shown in Table S6 (Supplementary Information). This result can be explained by the different optical properties of WS_2 and MoSe_2. At 600 nm excitation wavelength, the real part of the dielectric constant of MoSe_2 is 2 times as that of WS_2, which indicates MoSe_2 layers have higher energy.

Figure 2. The minimum reflectivity in SPR curve as a function of the number of WS_2 layers at 600 nm excitation wavelength with various thickness of the gold thin film and silicon nanosheet (a) 0 nm (b) 5 nm (c) 7 nm; and the corresponding changes in the resonance angle for a fixed refractive index change in the biomolecular analyte ($\Delta n_{bio} = 0.005$) as a function of the number of layers of WS_2 at 600 nm excitation wavelength with various thickness of gold thin film and silicon nanosheet (d) 0 nm (e) 5 nm (f) 7 nm.
absorption compared to WS₂ layers. As a result, the minR of silicon-MoSe₂ enhanced model is lower than that of silicon-WS₂. However, the absorbed energy is not completely transferred to enhance the evanescent field due to the energy loss during the process. It is known that a dielectric material with a large real part and a small image part of the dielectric function have low energy loss, therefore the WS₂ layers have much lower energy loss than that of MoSe₂. Furthermore, the penetration depth of the evanescent field in the biomolecular analyte layer of the silicon-WS₂ model is deeper than that of the silicon-MoSe₂ model, since the real part of dielectric constant of WS₂ is smaller than MoSe₂ [6]. Consequently, the evanescent field of silicon-WS₂ model is more sensitive than that of silicon-MoSe₂ model to the refractive index change in biomolecular analyte. All these factors contributed to higher sensitivity in the silicon-WS₂ enhanced scheme at 600 nm excitation wavelength. Similarly, the thickness combination for excitation wavelengths ranging from 633 nm to 1024 nm were also optimized (see Supplementary Information Tables S2–S20). Based on these results, we could conclude that although the MX₂ layers with large real values of the dielectric constant contributed to the increased energy absorption, the intrinsic energy loss in these layers played a more significant role in the sensitivity of the multi-layered system.

**Influence of excitation wavelength.** In order to obtain the best SPR sensing performance, it is also important to achieve a relatively low FWHM as it promises more accurate determination of the angular modulation. The value of the FWHM is mainly depends on two factors: excitation wavelength and the number of MX₂ layers. As reported previously, using longer excitation wavelength results in a narrower resonance curve⁶. As for the latter factor, our simulation results showed that large number of MX₂ layers generated a higher value...
of FWHM since the additional MX2 layers resulted in higher electron energy loss and reduced the accuracy. To investigate the effect of the excitation wavelength, we summarized the optimized thickness combinations of gold thin film and silicon nanosheet for five different excitation wavelengths. As shown in Table 1, three features were observed: (i) By using 600 nm, 633 nm, 660 nm and 785 nm excitation wavelengths, the best performances were all achieved in the silicon-WS2 sensing model except for the 1024 nm excitation wavelength which was attained in the silicon-MoS2 scheme. This was due to the hybrid effects of the energy absorption and energy loss of MX2 layers at the different excitation wavelengths. (ii) For the visible range, the optimized thickness of silicon was 7 nm and the WS2 layers were required to be ultra-thin (i.e., 1–3 layers), the predicted sensitivity were all above 140 Deg/RIU. However, for the near-infrared range (i.e., 785 nm and 1024 nm), the optimized thickness of silicon and gold were determined to be 5 nm and 40 nm respectively, with the optimized number of MX2 layers being at least 13 and above. (iii) It is worth noting that as the excitation wavelength redshifted, the optimized sensitivity decreased gradually. The dielectric constant changes significantly with the increasing incident wavelengths. The real part of the dielectric constant is related to the reflectivity of the interface while the imaginary part is indicative of the energy absorption. Therefore, according to Table 1, as the wavelengths progressed into the near-infrared region, the hybrid effects resulted in higher attenuation of the evanescent field and thus lower SPR sensitivity was observed49.

**Optimized scheme for each of the Silicon-MX2 model.** Finally, the parameters yielding the best performance for each of the silicon-WS2, silicon-WSe2, silicon-MoS2, silicon-MoSe2 enhanced SPR models are

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**Figure 4.** The minimum reflectivity in SPR curve as a function of the number of WS2 layers at 660 nm excitation wavelength with various thickness of the gold thin film and silicon nanosheet (a) 0 nm (b) 5 nm (c) 7 nm; and the corresponding changes in the resonance angle for a fixed refractive index change in the biomolecular analyte ($\Delta n_{bio} = 0.005$) as a function of the number of layers of WS2 at 660 nm excitation wavelength with various thickness of gold thin film and silicon nanosheet (d) 0 nm (e) 5 nm (f) 7 nm.
presented in Table 2. The optimized excitation wavelength and thickness of silicon nanosheet, gold thin film are the same for all the silicon-MX₂ enhanced models except for the silicon-WS₂ model. For the silicon-WS₂ scheme, it possessed the highest sensitivity of 155.68 Deg/RIU with 35 nm thick gold and 7 nm silicon at the 600 nm wavelength. The 3D plots shown in Fig. 7a–d further validates the optimized parameters by depicting the sensitivity as a function of number of MX₂ layers and gold thickness with the corresponding optimized silicon thickness and excitation wavelengths. From Fig. 7b–d, the highest value of sensitivity seemed to be attainable with 50 nm thick gold thin film and monolayer MX₂. However, the minimum SPR reflectivity under these condition were greater than 0.03, leading to low energy-transfer efficiencies. In addition, the slopes of the sensitivity as a function of MX₂ layers coated on 40 nm-thickness Au thin film were the sharpest among all the other Au thicknesses. Therefore, we concluded that the highest sensitivity of silicon-MoS₂, silicon-MoSe₂, and silicon-WSe₂ models were obtained with 40 nm thick gold films. We also compared the sensitivity of our optimized configuration with the well-known conventional Kretschmann configuration (with 50 nm thickness gold film). As shown in Fig. 8, the sensitivity of our optimized silicon-WS₂ enhanced configuration (155.68 Deg/RIU as shown in red solid line) is 3 times more sensitive than that of the Kretschmann design (53.40 Deg/RIU as shown in blue dashed line). For the Kretschmann configuration without any dielectric layers, the only absorbing medium in the system is the metal film; whereas in the silicon-WS₂ enhanced structure the additional silicon nanosheet and MX₂ layers can serve as the absorption medium as well. Moreover, the high refractive index of silicon nanosheet and the high real part of MX₂ dielectric constant can enhance the evanescent field at the metal interface. These double effects result in drastic sensitivity enhancement in the silicon-WS₂ enhanced structure. In order to

Figure 5. The minimum reflectivity in SPR curve as a function of the number of WS₂ layers at 785 nm excitation wavelength with various thickness of the gold thin film and silicon nanosheet (a) 0 nm (b) 5 nm (c) 7 nm; and the corresponding changes in the resonance angle for a fixed refractive index change in the biomolecular analyte (Δn_bio = 0.005) as a function of the number of layers of WS₂ at 785 nm excitation wavelength with various thickness of gold thin film and silicon nanosheet (d) 0 nm (e) 5 nm (f) 7 nm.
demonstrate the validity of our N-layer 2D models, we have experimentally tested the SPR sensing ability with 3-layer graphene-coated Au thin film. And both the angular and phase measurement results matched well with our theoretical analyses (see Supplementary Figs S1–S3).

Figure 6. The minimum reflectivity in SPR curve as a function of the number of WS₂ layers at 1024 nm excitation wavelength with various thickness of the gold thin film and silicon nanosheet (a) 0 nm (b) 5 nm (c) 7 nm; and the corresponding changes in the resonance angle for a fixed refractive index change in the biomolecular analyte ($\Delta n_{biol} = 0.005$) as a function of the number of layers of WS₂ at 1024 nm excitation wavelength with various thickness of gold thin film and silicon nanosheet (d) 0 nm (e) 5 nm (f) 7 nm.

| Excitation wavelength (nm) | Type of TMDC | Gold thickness (nm) | Silicon thickness (nm) | Number of WS₂ layers (L) | Minimum Reflectivity | $\Delta\theta_{SPR}$ (Deg) ($\Delta n_{biol} = 0.005$) | Sensitivity (Deg/RIU) | FWHM (Deg) |
|---------------------------|--------------|--------------------|-----------------------|--------------------------|---------------------|-----------------------------------------------|---------------------|-------------|
| 600                       | WS₂          | 35                 | 7                     | 1                        | 2.5592 $\times 10^{-2}$ | 0.7784                          | 155.68              | 17.4644     |
| 633                       | WS₂          | 40                 | 7                     | 1                        | 2.4099 $\times 10^{-2}$ | 0.7394                          | 147.88              | 16.2417     |
| 660                       | WS₂          | 40                 | 7                     | 3                        | 3.3778 $\times 10^{-3}$ | 0.7282                          | 145.64              | 16.8245     |
| 785                       | WS₂          | 40                 | 5                     | 13                       | 2.5767 $\times 10^{-3}$ | 0.6395                          | 127.90              | 15.5232     |
| 1024                      | MoS₂         | 40                 | 5                     | 26                       | 1.6573 $\times 10^{-3}$ | 0.4499                          | 89.98               | 17.5445     |

Table 1. The optimized values of gold thin film, silicon nanosheet thickness and the number of MX₂ layers with corresponding changes in resonance angle, sensitivities and FWHMs in SPR curves for 600 nm, 633 nm, 660 nm, 785 nm and 1024 nm excitation wavelengths.
Conclusion

In this study, we demonstrated a silicon nanosheet and 2D MX\textsubscript{2} enhanced surface plasmon resonance biosensor. Based on the Kretschmann configuration, the system consists of SF10 triangular prism, gold thin film, silicon nanosheet and 2D MX\textsubscript{2} film (MoS\textsubscript{2}/MoSe\textsubscript{2}/WS\textsubscript{2}/WSe\textsubscript{2}). To investigate the enhancement effect of each MX\textsubscript{2} materials, we designed four enhanced models, namely, silicon-WS\textsubscript{2}, silicon-WSe\textsubscript{2}, silicon-MoS\textsubscript{2} and silicon-MoSe\textsubscript{2}. Maxwell’s equations, Fresnel equations and transfer matrix method were used to analyze the change in resonance angle and the corresponding sensitivity for angular modulation. To study the influence of the excitation wavelengths to sensing performance, we studied five different excitation wavelengths, namely, 600 nm, 633 nm, 660 nm, 785 nm and 1024 nm. The results showed that the silicon nanosheet together with each of the four types of 2D MX\textsubscript{2} layers could compensate the SPR effect of gold and significantly improve the sensitivity of the biosensor. However, excessive MX\textsubscript{2} layers would result in increased energy loss and reduced the sensitivity. Therefore, in order to optimize the sensitivity, the thickness of gold film, silicon nanosheet and MX\textsubscript{2} layers were investigated to minimize the reflectance and width of SPR curve to achieve a system with higher angular scanning accuracy.

The combination of the optimized parameters for each excitation wavelength and each silicon-MX\textsubscript{2} nanosheet enhanced model were also presented with the highest SPR sensitivity of 155.68 Deg/RIU achieved using 35 nm thick gold film, 7 nm thick silicon nanosheet and a monolayer WS\textsubscript{2} under the illumination of a 600 nm excitation wavelength source.
Methods

The wavelength-dependent refractive index for each layer. In this study, the theoretical analyses were performed with excitation light wavelengths in visible and near infrared ranges. Five wavelengths were chosen as 600 nm, 633 nm, 660 nm, 785 nm and 1024 nm respectively. The proposed configuration consists of 6 layer components, namely the SF10 prism, gold thin film, silicon nanosheet, 2D MX2 nanolayers (MoS2/WS2/MoSe2/WSe2), biomolecular analyte layer and sensing medium layer. The first component is the SF10 prism whose refractive index is given by:

\[
n_{\text{prism}}(\lambda) = \left( \frac{1.62153902 \lambda^2}{\lambda^2 - 0.0122241457} + \frac{0.256287842 \lambda^2}{\lambda^2 - 0.0595736775} + \frac{1.64447552}{\lambda^2 - 147.468793} + 1 \right)^{1/2}
\]

where \( \lambda \) represents the wavelength of the light source in \( \mu m \) and Eq. (2) is valid for wavelengths from 0.38 \( \mu m \) to 2.5 \( \mu m \). The refractive index of the gold thin film is determined using the Drude model by:

\[
n_{\text{gold}}(\lambda) = \left( \varepsilon'_{\text{gold}} + i \varepsilon''_{\text{gold}} \right)^{1/2} = 1 - \frac{\lambda^2 \lambda_p}{\lambda^2 - \lambda_c^2 + i \lambda^2}
\]

where \( \lambda_p (1.6826 \times 10^{-7} m) \) denotes the plasma wavelength and \( \lambda_c (8.9342 \times 10^{-6} m) \) is the collision wavelength. \( \lambda \) in Eq. (3) represents the wavelength of the light source in \( \mu m \), which is valid from 0.18 \( \mu m \) to 1.94 \( \mu m \). The thickness of the gold thin film was varied from 30 nm to 50 nm in this study. The refractive index of the silicon nanosheet is calculated by:

\[
n_{\text{silicon}}(\lambda) = A + A_1 e^{-\lambda t_1} + A_2 e^{-\lambda t_2}
\]

where \( A = 3.44904, A_1 = 2271.88813, A_2 = 3.39538, t_1 = 0.058304 \) and \( t_2 = 0.30384 \), and \( \lambda \) is the wavelength in \( \mu m \). The thickness of the silicon sheet was varied from 0 nm to 7 nm. The fourth layer is the core functional layer – 2D MX2 (MoS2/WS2/MoSe2/WSe2) whose refractive indexs and the monolayer thickness \( t_{MX2} \) are summarised in Table 3. The thickness of MX2 layers \( d_{MX2} \) is described by \( d_{MX2} = t_{MX2} \times L \), where \( L \) represents the number of MX2 layers. The fifth layer is the biomolecular analyte layer which dissolved in sensing medium. The change in refractive index of this layer can be induced by absorption of the biomolecules on the MX2 surface. Here the refractive index of this layer can be represented as \( n_{\text{analyte}} = 1.33 + \Delta n_{\text{bio}} \), where \( \Delta n_{\text{bio}} \) indicates the change in refractive index of the biomolecular analyte. The thickness of this layer \( d_{\text{analyte}} \) is fixed at 100 nm. The last layer is the sensing medium, the refractive index is defined as \( n_{\text{solvent}} = 1.33 \).

Reflection coefficient \( (r_p) \) and Reflectivity \( (R_p) \). In this study, the SPR biosensor model is based on the well-known Kretschmann's attenuated total reflection (ATR) structure. The incident light passes through the prism and is totally reflected at the base of the prism. Most of the light energy is absorbed by the metal and dielectric layers to generate the evanescent wave which propagates along the interface and penetrates into the sensing medium. When the evanescent wave vector \( k_e \) matches with the surface plasmon wave vector \( k_p \), the resonance occurs as shown in Eq. (1). In order to analyse our N-layer structure model, the transfer matrix method (TMM) was employed. Each layer in our system was stacked horizontally in z-axis direction. All layers are defined by the parameters \( n_k, \varepsilon_k \) and \( d_k \), which represent the refractive index, dielectric constant and thickness of the \( k \)th layer, respectively. \( \theta_i \) denotes the incident angle of the \( k_{th} \) layer and \( \lambda \) stands for the excitation wavelength.
The extinction coefficient refers to the mass attenuation coefficient. The imaginary part \( \epsilon'' \) relates to the dissipation of energy within the medium. (The complex refractive index \( \epsilon = \epsilon' + \epsilon'' i \), where the real part \( \epsilon' \) relates to the stored energy within the medium and the imaginary part \( \epsilon'' \) relates to the dissipation of energy within the medium. (The complex refractive index \( \epsilon \) is defined as \( \epsilon = \sqrt{\epsilon_r} = n + k \), where the real part \( n \) indicates the phase velocity, while the imaginary part \( k \) known as the extinction coefficient refers to the mass attenuation coefficient).

To obtain the reflected intensity, Fresnel equations and relevant boundary conditions were introduced. In the calculations, the first boundary of the tangential fields was assumed as \( Z_1 = 0 \), and the tangential fields at the last boundary \( Z_{N-1} \), there by giving rise to Eq. (5) as follows:

\[
\begin{bmatrix}
U_1 \\
V_1
\end{bmatrix}
= M
\begin{bmatrix}
U_{N-1} \\
V_{N-1}
\end{bmatrix}
\]  

(5)

where \( U_1, U_{N-1} \) represent the tangential components of the first and the last layers in the electric fields, while \( V_1, V_{N-1} \) denote the corresponding components in magnetic fields. \( M \) refers to the characteristic matrix of the \( N \)-layer model. For the \( p \)-polarized light, the characteristics matrix is given by:

\[
M = \prod_{k=2}^{N-1} M_k = \begin{bmatrix} M_{11} & M_{12} \\ M_{21} & M_{22} \end{bmatrix}
\]

(6)

With

\[
M_k = \begin{bmatrix} \cos \beta_k & (-i \sin \beta_k)/q_k \\ -iq_k \sin \beta_k & \cos \beta_k \end{bmatrix}
\]

(7)

Where

\[
q_k = (\varepsilon_k - n_k^2 \sin^2 \theta_k)^{1/2} = \frac{\lambda \beta_k}{2 \pi d_k \varepsilon_k}
\]

(8)

\[
\beta_k = \frac{2 \pi n_k \cos \theta_k(Z_k - Z_{k-1})}{\lambda}
\]

(9)

Thus, the four elements \( M_{11}, M_{12}, M_{21} \) and \( M_{22} \) of the matrix \( M \) can be calculated. According to the Fresnel's equations, the complex reflection coefficient \( r_p \) of \( p \)-polarized incident electromagnetic field can be described by:

\[
r_p = \frac{(M_{11} + M_{12}q_N)q_1 - (M_{21} + M_{22}q_N)}{(M_{11} + M_{12}q_N)q_1 + (M_{21} + M_{22}q_N)}
\]

(10)

where, \( q_1 \) and \( q_N \) can be calculated from Eq. (8), which represent the relative components of the first layer and the \( N \)-th layer respectively. Therefore, the system reflectivity \( R_p \) for the \( p \)-polarized incident light can then be obtained by taking the square of the reflection coefficient \( r_p \) shown as follows:

\[
R_p = |r_p|^2
\]

(11)

The angular sensitivity is defined as the ratio between the change of the resonance angle to the change of the analyte refractive index \( n_a \) in Eq. (12).

\[
S \equiv \frac{d\theta_{\text{SPR}}}{dn_a}
\]

(12)

It can be simplified to

| TMDC | \( T_{\text{st}} \) (nm) | \( \lambda = 600 \text{ nm} \) | \( \lambda = 660 \text{ nm} \) | \( \lambda = 785 \text{ nm} \) | \( \lambda = 1024 \text{ nm} \) |
|-------|------------------|------------------|------------------|------------------|------------------|
| MoS\(_2\) | 0.65 | \( \epsilon' = 17.7967 \) | \( \epsilon' = 24.4368 \) | \( \epsilon' = 23.4129 \) | \( \epsilon' = 21.4675 \) | \( \epsilon' = 20.8372 \) |
| MoSe\(_2\) | 0.70 | \( \epsilon' = 21.3204 \) | \( \epsilon' = 20.3560 \) | \( \epsilon' = 19.3366 \) | \( \epsilon' = 17.7994 \) | \( \epsilon' = 14.9028 \) |
| WS\(_2\) | 0.80 | \( \epsilon' = 12.0258 \) | \( \epsilon' = 23.8511 \) | \( \epsilon' = 19.9701 \) | \( \epsilon' = 16.0968 \) | \( \epsilon' = 25.0445 \) |
| WSe\(_2\) | 0.70 | \( \epsilon' = 19.4125 \) | \( \epsilon' = 20.5156 \) | \( \epsilon' = 18.7344 \) | \( \epsilon' = 19.0563 \) | \( \epsilon' = 23.6474 \) |

Table 3. The optical constants of 2D MoS\(_2\), MoSe\(_2\), WS\(_2\), and WSe\(_2\) with different excitation wavelengths from 600 nm to 1024 nm. The dielectric constant is described by \( \epsilon' = \epsilon'' i \), where the real part \( \epsilon' \) relates to the stored energy within the medium and the imaginary part \( \epsilon'' \) relates to the dissipation of energy within the medium.
The sensitivity describes the change in the optical signal (i.e., changes in the resonance angle) corresponding to the minute changes in the refractive index of the biomolecular analyte. The sensitivity depends on the evanescent field strength which directly related to the absorbed light energy. Another key parameter of interest is the width of the SPR curve, which related to the accuracy of the sensing system. The width of SPR curve is determined by the dielectric function of the metallic silicon-MX2 thin film. Generally, a large value of $\frac{\Delta \theta_{\text{max}}}{\Delta n}$ results in a narrow resonance curve. In this work, the full width at half maximum (FWHM) is calculated to investigate the SPR curve width, as shown in Eq. (14).

$$FWMH = \frac{1}{2} \times (\theta_{\text{min}} + \theta_{\text{max}})$$

(14)

where $\theta_{\text{min}}$ is the incident angle corresponding to the minimum reflected intensity, and $\theta_{\text{max}}$ is the incident angle with the maximum reflected intensity base on the SPR curve.

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Author Contributions

Q.O., S.Z. and K.-T.Y. conceived the idea. Q.O., S.Z., L.J., L.H., G.X. and J.Q. designed and carried out the research. Q.O., S.Z. and K.-T.Y. co-wrote the manuscript. X.-Q.D., J.Q., S.H. and P.C. provided suggestions, and all authors commented on the manuscript.

Additional Information

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