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Tuning micropillar cavity birefringence by laser induced surface defects

Cristian Bonato,1,a) Dapeng Ding,1 Jan Gudat,1 Susanna Thon,2 Hyochul Kim,2 Pierre M. Petroff,2 Martin P. van Exter,1 and Dirk Bouwmeeste1,2

1Huygens Laboratory, Leiden University, P.O. Box 9504, 2300 RA Leiden, The Netherlands
2University of California Santa Barbara, Santa Barbara, California 93106, USA

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We demonstrate a technique to tune the optical properties of micropillar cavities by creating small defects on the sample surface near the cavity region with an intense focused laser beam. Such defects modify strain in the structure, changing the birefringence in a controllable way. We apply the technique to make the fundamental cavity mode polarization-degenerate and to fine tune the overall mode frequencies, as needed for applications in quantum information science. © 2009 American Institute of Physics. [doi:10.1063/1.3276550]

Much work has been recently devoted to the development of semiconductor optical microcavities1 for quantum information processing applications. For example, a carefully designed cavity can be used to tailor the properties of single photon sources and to maximize their yield.2–4 In addition, quantum dots coupled to semiconductor microcavities provide a very promising system for the implementation of cavity quantum electrodynamics experiments,5,6 and for hybrid quantum information protocols in which photons are used for long-distance transmission and matter qubits for local storage and processing.7,8 However, some technical issues are yet to be solved: among these, the fine tuning of the microcavity optical properties. To generate quantum superpositions and to exploit quantum interference effects, which are at the heart of quantum information protocols, the states which form the superposition must be indistinguishable. In other words, if the polarization degree of freedom encodes the quantum bit, there must be no way to obtain information about its polarization by observing other degrees of freedom. Therefore, the implementation of a quantum interface between the polarization state of a single photon and a two-level system requires the cavity mode to be polarization-degenerate. A second problem is that the cavity resonance frequency and the frequency of the two-level system, in our case semiconductor self-assembled quantum dots, must be matched with a precision which is currently impossible to obtain deterministically in the fabrication process. Several frequency tuning techniques are commonly used (like temperature tuning5 or Stark shift9) but they are temporary and leave the cavity birefringence unchanged.

Here we demonstrate an all-optical technique, originally developed to tailor the polarization properties of vertical-cavity surface-emitting lasers,10,11 to apply a controlled and permanent birefringence to the optical micropillar cavities. In this way the frequency shift of the two polarization modes can be tuned at will, allowing polarization-degeneracy. We will show that this technique permits control of the frequencies of the two polarization modes almost independently of one another, providing a tuning range of a few Angstroms.

The technique is based on the creation of a permanent defect on the surface of the sample near the cavity by means of a strongly focused laser beam (see Fig. 1, upper figure). The sample locally melts, creating a hole with some material accumulated on the edges. Such holes are 3–5 μm wide, with depth varying from 30 nm to 2 μm depending on the burning time and affect the strain (and therefore the birefringence) in the structure. The magnitude of the induced stress can be varied by tuning the laser power and the exposure time, while its orientation is determined by the position of the burn around the cavity. In our experiments, the defect is created by a Ti:sapphire laser (about 250 mW power) tuned to 770 nm in order to have sufficient absorption by the semiconductor material, tightly focused on the structure by a high numerical aperture (NA) aspheric lens L1 (focal length f0 = 4.02 mm, NA = 0.6). The burn is precisely positioned onto the sample by means of an optical system consisting of the focusing lens L1 and a second lens L2 (focal length f’ = 150 mm) which images the sample onto a charge-coupled device (CCD) camera (placed in the focal plane of the lens L2).

a)Electronic mail: bonato@molphys.leidenuniv.nl.

FIG. 1. (Color online) (a) Hole burning locations for polarization-splitting compensation. (b) Atomic force microscope image of a hole. (c) Compensation of the polarization splitting for the fundamental cavity mode.
The micropillar samples we investigated were grown by molecular-beam epitaxy on a GaAs substrate. The microcavity consists of two distributed Bragg reflector (DBR) mirrors made by alternating layers of GaAs and Al0.3Ga0.7As (one-quarter optical thickness, 32 pairs for the bottom DBR, 4.8 μm thick, and 23 pairs for the top DBR, 3.5 μm thick), spaced by a λ-thick GaAs cavity layer with embedded InGaAs/GaAs self-assembled quantum dots. Trenches are etched through the sample (4.3 μm thick, down through the active region), and the sample is placed in an oxidation furnace to create an oxidation aperture in the AlAs layer which provides gentle lateral confinement of the optical mode. The micropillar structures, typically 30 μm in diameter, are very robust (see Ref. 4 for a three-dimensional sketch). The oxidation aperture determines a mode waist of about 1–2 μm at the center of the structure. In this letter, we will focus on the properties of the fundamental transverse cavity mode, which exhibits a very good spatial Gaussian shape and is split into two orthogonally polarized submodes (M_A(00) and M_B(00)). The defects, burnt a few microns away from the center of the micropillar, do not reduce the optical quality of the cavity.

The spectrum of the cavity modes can be characterized by pumping the semiconductor material above the bandgap with a Ti-Sapphire beam of a few mW and observing the cavity-shaped photoluminescence on a spectrometer (resolution 5.5 GHz/pixel) equipped with a CCD array. Its polarization dependence is characterized by placing an analyzer consisting of a fixed linear polarizer and a rotating half-wave plate in front of the spectrometer, so that the polarization state in the spectrometer is constant and the measurements are not affected by the polarization response of the grating.

The spectral splitting of the two polarization modes is typically around a few GHz, which is smaller than the spectral width of the mode and comparable to the spectrometer resolution. A direct measurement of the centers of the two overlapping modes is therefore not possible. However, the two modes are orthogonally polarized, so that M_A(00) (M_B(00)), centered at frequency ν_A (ν_B) dominates completely at some analyzer angle θ_A (θ_A + π/2). Rotating the analyzer, the center of the peaks shifts periodically between ν_A and ν_B. The accuracy in the determination of the position of the peak can be greatly enhanced beyond the spectrometer resolution by means of a Lorentzian fit of the peak. The periodic oscillations of the peak center as a function of the analyzer angle can be clearly resolved, as shown in Fig. 1. Experimental data for a cavity with no holes burnt are shown by the blue curve: the separation between the central frequencies of the two polarization modes is measured to be Δν = 13.7 ± 0.3 GHz. The full width at half maximum of the fundamental mode peak, measured selecting one single polarization mode is 30.1 ± 0.4 GHz.

Next we modified the cavity by burning holes at locations chosen on the basis of the expected angular and 1/r dependence of the induced strain (r being the distance of the burn from the cavity center) and of initial tests performed on different cavities. After burning one hole for one minute on the surface between the southern and western trenches [hole (1) on the inset in Fig. 1], close to the cavity center, the splitting is reduced from 13.7 ± 0.3 to 6.1 ± 0.7 GHz. A second hole (2) did not lead to a strong reduction. After burning two more holes on the edge of the eastern trench [holes (3) and (4), each one for one minute], the splitting is reduced to Δν = 2.0 ± 0.2 GHz.

More tests were performed on a new cavity in order to understand how the spectrum of the two peaks changes when a hole is burnt at different positions (see Fig. 2). The results are illustrated in Fig. 3, where the central wavelengths of the two polarization peaks (λ_A and λ_B) are plotted on the horizontal and vertical axes. Each point on the graph corresponds to a hole burnt on the sample. First, eight holes were burnt between the southern and western trenches, starting farther from and moving closer to the cavity centers, and then six more holes were burnt between the northern and the eastern trenches continuing in the same direction (along the [001] crystal lattice orientation). The effect of the first few holes is to reduce the spectral splitting of the two polarization modes, reaching a minimum of Δν = 3.2 ± 0.4 GHz. After that, λ_B remains approximately constant, while λ_A increases. The closer the holes are burnt with respect to the center, the larger the frequency shift. Due to the combined effect of the 14 holes burnt along this direction, Δν becomes 108 ± 1 GHz and the two peaks can be clearly resolved (inset in Fig. 3). Burning holes along the orthogonal direction, λ_B changes much faster than λ_A. By adjusting the hole orientation and distance from the center, we have a way to tune almost in-
dependently the frequency of the two polarization modes with a range of about 100–150 GHz.

This effect can be explained by a simple model, based on the tensorial relationship between stress, strain, and the optical properties of the material.\textsuperscript{14} Essentially, the anisotropic component of the stress changes the splitting and the isotropic component affects the absolute frequencies of the two submodes. In particular, we find that for $N$ holes burnt along the $x$ direction

$$n_{B}^{(N)} = n_0 + (n_A^{(N)} - n_0) \left( -\frac{\Pi_2}{\Pi_1} \right),$$

which corresponds to a straight line with slope $\rho_1 = -\Pi_2/\Pi_1$. $\Pi_1$ and $\Pi_2$ are quantities which depend on the tensorial elastic ($C_{ij}$) and elasto-optic ($\rho_{ij}$) coefficients of the material ($\Pi_1 = p_{11}C_{11} - p_{12}C_{12}$ and $\Pi_2 = p_{11}C_{12} - p_{12}C_{11}$). For the holes burnt along the $y$ direction we find a similar linear relationship with inverted slope $\rho_2 = -\Pi_1/\Pi_2$. As can be seen in Fig. 3, the data fit quite well with this model: the fitted slope for the lines is $4.0 \pm 0.5$. Literature values for the bulk GaAs and AlAs thermal and elasto-optic properties\textsuperscript{15} give a slope of around 1.5. However, we do not expect these values to be perfectly compatible since our model does not take into account the bimorphic structure formed by the oxidized AlAs layer and by the DBR mirrors.

The cavity resonance frequency and polarization splitting for holes burnt at room temperature were measured at 4 K, showing values significantly different from the room-temperature ones. We repeated the hole-burning process directly at low-temperature, increasing the burning laser power to 500 mW (532 nm). Polarization degeneracy could be achieved as well as frequency tuning, albeit with a different slope $\rho_2 = 1.2 \pm 0.5$. The stability of the effects was tested by warming up and cooling down the device a few times: a difference of the order of 10% was found for the first cooldown after burning (consistent with the results in Ref. 11), while the deviation in the splitting is within 1–2 GHz for the successive cooldowns.

In conclusion, we introduced a technique to permanently tune the polarization and spectral properties of optical micro-pillar cavities. By laser-burning a small defect on the sample surface near the cavity, we can induce a controllable amount of birefringence in the structure. By adjusting the position of the defect, we control the central wavelengths of the two polarization submodes of the fundamental cavity mode. This technique enables the implementation of polarization-degenerate semiconductor micropillars for quantum information processing and it may find applications for fine tuning of other kinds of semiconductor microcavities whose optical properties are influenced by material strain, such as photonic crystal defect cavities and microdisk cavities.

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