A Novel Metastable Pentavalent Plutonium Solid Phase on the Pathway from Aqueous Plutonium(VI) to PuO$_2$ Nanoparticles

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Abstract: Here we provide evidence that the formation of PuO$_2$ nanoparticles from oxidized Pu$^{VI}$ under alkaline conditions proceeds through the formation of an intermediate Pu$^V$ solid phase, similar to NH$_2$PuO$_4$CO$_3$, which is stable over a period of several months. For the first time, state-of-the-art experiments at Pu M, and at L$_3$ absorption edges combined with theoretical calculations unambiguously allow to determine the oxidation state and the local structure of this intermediate phase.

Plutonium plays a prominent role in nuclear energy production but nuclear accidents and nuclear weapons tests have led to the release of Pu and other hazardous isotopes into the environment in the past, and Pu contamination has been detected in waters and soils.[1] Based on such cases, several countries decided to shut down the operation of the oldest nuclear facilities and put effort into improving the safety of nuclear waste storage in order to prevent further release of radioactive nuclides into the environment. To progress in this direction, it is fundamental to deepen our basic knowledge of the chemistry of actinides in environmentally relevant conditions by making compounds, characterizing them, and understand them experimentally and theoretically. Thanks to the increased experimental sensitivity, recent cross-activities between theory and experiment, and different synthetic approaches, such a goal becomes reachable.

In spite of the low solubility of the most prevalent environmental species, Pu has been shown to be transported by groundwater from contaminated sites for several kilometers in the form of colloids, with Pu being absorbed on clays,[3] iron oxides,[3] or natural organic matter.[4] In the near-field conditions of geological repositories of spent nuclear fuel and other radioactive wastes, the formation of intrinsic PuO$_2$ colloids is a key scenario.[5] Therefore, the characterization of such intrinsic colloidal nanoparticles (NPs) in aqueous solution has recently received much attention.[6–10] The most debated question is the structural nature of these NPs (crystalline vs. amorphous) as well as the presence of Pu$^V$ and other oxidation states in small NPs (<3 nm).[9,11–15] Various studies used different synthetic approaches and different solution conditions to examine a precipitated product, either amorphous or crystalline. This has led to a controversy which has not been resolved. For example, Walther et al.[14] observed evidence for multiple Pu oxidation states (III, IV, V) in the early stages of hydrolysis and polymerization of PuO$_2$ colloids at pH 0.5–1.0, while Rothe et al.[9] reported Pu$^IV$ oxyhydroxide-colloid formation. Conradson et al.[15] examined solid precipitates prepared by a variety of synthetic approaches and argued for the presence of Pu$^V$ in nonstoichiometric PuO$_{1-x}$ solids.

One of the most fundamental properties of the chemical behavior of Pu is the variety of its oxidation states. The oxidation state is defined by the number of electrons that are removed from the valence orbitals of a neutral atom. In the pentavalent oxidation state, Pu has three electrons in the 5f shell, leaving the 6d orbitals empty. The oxidation state of Pu determines its chemical behavior and reactivity. Four oxidation states (from III to VI) may co-exist under environmental conditions, while oxidation states VII and VIII are proposed to be stable under highly alkaline oxidative conditions.[16] Oxidation states of aqueous, solid-state, and interfacial Pu

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Figure 1. a) Experimental HERFD data at the Pu M\(_4\) edge from two plutonium phases obtained during the synthesis of PuO\(_2\) nanoparticles (NPs) from a Pu\(^{VI}\) precursor at pH 11. Blue curve: spectrum of the intermediate Pu\(^{IV}\) solid phase appearing during the synthesis of the PuO\(_2\) NPs; red curve: spectrum of the final phase of PuO\(_2\) NPs. The spectra of a PuO\(_2\) bulk sample (grey curve) and of KP\(_2\)O\(_3\) (green curve) are also shown as references for Pu\(^{VI}\) and Pu\(^{IV}\) oxidation states, respectively. Data were collected with an X-ray emission spectrometer set to the maximum of the M\(_{\beta}\) emission line at 3534 eV. b) Experimental HERFD spectra of PuO\(_2\) and KP\(_2\)O\(_3\) compared with the results of Anderson impurity model calculations.

species have been previously determined using Pu L\(_3\) edge\(^{6,7,17}\) X-ray absorption near edge structure (XANES) spectroscopy. The Pu edge of the L\(_3\) XANES spectrum of Pu\(^V\) always shows a characteristic energy shift towards low energies compared to Pu\(^{IV}\) and Pu\(^{VI}\) XANES spectra. The experimental energy resolution of the recorded XANES data can be improved if the spectra are recorded in the high energy resolution fluorescence detection (HERFD) mode.\(^{8}\) Nevertheless, at the Pu L\(_3\) edge, the electrons are excited from the 2p core level to the 6d level, which is always unoccupied independent of the Pu oxidation state. For uranium systems, we have previously shown that HERFD experiments at the U M\(_4\) edge\(^{18-20}\) are much more informative on the oxidation state and electronic structure than measurements at the L edges. X-ray absorption at the M\(_4\) edge of actinides probes 5f states via transitions from the 3d core level. To our knowledge, HERFD data at the Pu M\(_4\) edge have never been reported in the literature and have never been exploited.

Figure 1a shows the first experimental HERFD data at the Pu M\(_4\) edge for the PuO\(_2\) and KP\(_2\)O\(_3\)(CO\(_3\)) (solid) systems with Pu\(^{IV}\) and Pu\(^{V}\) oxidation states, respectively. Data were collected with an X-ray emission spectrometer set to the maximum of the M\(_{\beta}\) emission line at 3534 eV. Synthesis procedures and the characterization of both materials are reported in the Supporting Information. The HERFD spectrum of PuO\(_2\) clearly shows two intense peaks, at \(\approx 3970.2\) eV and \(\approx 3971.8\) eV. According to the results of calculations carried out in the framework of the Anderson impurity model (AIM; Figure 1b),\(^{22-24}\) the intensity and energy of these two peaks are a result of multiple factors, such as the strength of the intra-atomic and crystal-field interactions, and the degree of the Pu 5f\(\text{ligand } 2p\) hybridization in the ground and final states of the spectroscopic process. In comparison with PuO\(_2\), the HERFD spectrum of KP\(_2\)O\(_3\)(CO\(_3\)) shifts towards higher incident energies and shows a narrow profile with an asymmetric shape and a shoulder at the higher incident energy side. The results of the AIM calculations reported in Figure 1b show a good agreement with the experimental KP\(_2\)O\(_3\)(CO\(_3\)) HERFD spectrum, confirming the presence of the pentavalent Pu oxidation state in KP\(_2\)O\(_3\)(CO\(_3\)).

Due to dipole selection rules \((J = 0; \pm 1)\), the shape of the Pu M\(_4\) and M\(_5\) HERFD transitions is expected to be different. At the Pu M\(_4\) edge, the unoccupied 5f electronic levels with \(J = 5/2\) and \(7/2\) can be reached by an electron excited from the Pu 3d\(_{5/2}\) state, whereas only the \(J = 5/2\) state can be reached at the Pu M\(_5\) edge.\(^{22}\) A comparison between Pu M\(_4\) and Pu M\(_5\) spectra for several Pu systems is shown in Figure S1 (Supporting Information). The energy shifts between Pu\(^{IV}\), Pu\(^{IV}\), and Pu\(^{V}\) in solid compounds are found to be in the order of 2 eV (between Pu\(^{III}\) and Pu\(^{IV}\)) and 0.4 eV between Pu\(^{IV}\) and Pu\(^{V}\) (Table S1). A correct determination of the Pu oxidation state therefore requires the improved energy resolution of the absorption spectra provided by HERFD.

Figure 1a shows experimental HERFD data recorded at different stages during the synthesis of PuO\(_2\) NPs from the aqueous Pu\(^{IV}\) precursor. For this purpose, a solution of Pu\(^{IV}\) was added to an excess of ammonia. The measured pH value of the solution was 11. We kinetically traced the route of the Pu\(^{IV}\)-to-PuO\(_2\) transformation as a two-step process: during the first minutes, we observed the formation of an intermediate Pu phase consisting of yellow sludge (see Figure 2). Later, during the formation of PuO\(_2\) NPs, the intermediate phase dissolved and a different equilibrium phase (named “final phase” in the following) was formed.\(^{26}\) The Pu M\(_4\) HERFD spectrum recorded at the intermediate stage of the reaction is represented by the blue curve in Figure 1a. The spectrum clearly indicates the presence of the Pu\(^{V}\) oxidation state. This is supported by the good correspondence between energy and the relative intensities of the main features of the Pu M\(_4\) edge spectrum for KP\(_2\)O\(_3\)(CO\(_3\)) (green curve) and the Pu intermediate phase (blue curve).

Furthermore, the HERFD spectrum of the final product of the reaction, formed after 3 weeks of the precipitation
reaction, shows an identical profile to the one detected for PuO$_2$ single crystal, confirming that the reaction terminates with the formation of PuO$_2$ NPs with cubic structure and with the Pu$^{IV}$ oxidation state, as reported by Soderholm et al.[15] for Pu$_3$ clusters (Li$_3$(H$_2$O)$_6$[Pu$_3$O$_6$Cl$_6$(H$_2$O)$_3$]) isolated from the initially alkaline peroxide solution.[15]

The experimental data collected for the intermediate phase during the PuO$_2$ NPs synthesis show evidence of the Pu$^{V}$ oxidation state. The exact contribution of the different chemical states in the Pu M$_4$ HERFD data reported in Figure 1a was estimated by the ITFA program.[27] The results indicate that the spectrum of the intermediate Pu phase contains 87% of Pu$^{V}$ and 13% of Pu$^{IV}$ (with an estimated root-mean-square error of less than 2%, see Figure S2). We did neither observe a significant contribution of Pu$^{V}$ in the final phase (after the PuO$_2$ NPs were formed) nor a quantifiable amount of Pu$^{VI}$ (Table S2). The absence of Pu$^{V}$ in the final phase and the 100% presence of the Pu$^{IV}$ oxidation state after the PuO$_2$ NPs formation is an important result. At the same time, our data demonstrate that Pu$^{V}$-to-Pu$^{IV}$ reduction does not occur in a single step.[23] The Pu$^{IV}$ is first reduced to Pu$^{V}$ and then to Pu$^{IV}$.

Moreover, additional HERFD and EXAFS (extended X-ray absorption fine structure) experiments at the Pu L$_3$ edge gave us the opportunity to identify the intermediate phase forming in the course of the PuO$_2$ NPs growth. Figure 3 shows the comparison of the Pu HERFD L$_3$ edge data recorded for PuO$_2$ and the intermediate Pu phase during the PuO$_2$ NPs formation. As discussed previously, the L$_3$ spectrum of Pu$^{IV}$ compounds always shows a very characteristic energy shift towards low energies and a decrease of the L$_3$ white line intensity compared to Pu$^{III}$ and Pu$^{VI}$ systems[26,28,29] (Figure S3). The chemical shift of the intermediate Pu phase is clearly resolved in the HERFD data reported in Figure 3 and indicates the presence of the Pu$^{IV}$ oxidation state, in agreement with the Pu M$_4$ HERFD results. However, for actinide systems, HERFD at the L$_3$ edge is not as sensitive as the M$_4$ edge HERFD to the presence of minor contributions (<10%) from different oxidation states[28,29] HERFD at the L$_3$ edge is, however, extremely sensitive to the local structure around the absorber, which results in specific post-edge features.[10,30,31]

Ab-initio calculations on different structures were used to identify the intermediate Pu phase during the synthesis of the PuO$_2$ NPs. We simulated the HERFD spectra of several compounds containing Pu (Figures S4 and S5) in order to determine the Pu speciation of the intermediate Pu phase structure. The best agreement is found for NH$_4$PuO$_2$CO$_3$ in which Pu is present in the pentavalent state. The HERFD spectral shape reflects the d-density of states (DOS) of Pu apart from the small shoulder at the absorption edge, which is barely visible in the data but well resolved in the simulation and represents the Pu 5f DOS (Figure S6). The Pu d-DOS is involved in the bonds with O, C, and N. The Pu L$_3$ EXAFS data confirmed that the intermediate Pu phase formed during the PuO$_2$ NPs synthesis is compatible with NH$_4$PuO$_2$CO$_3$. Furthermore, the EXAFS spectrum (Figure S7 and Table S3) could be fitted with a model based on the crystal structure of NH$_4$PuO$_2$CO$_3$ that was previously published.[32] The fitted Pu–O distance of the triple-bond group is 1.82 Å, in good agreement with previously determined distances of 1.80–1.81 Å for Pu$^{IV}$ compounds[17,30] (see Table S4), while the crystallographic distance of 1.93 Å is most likely biased by the very weak scattering of oxygen in comparison to Pu$^{III}$.[33] We also found that the Pu–Pu coordination number in the experimental EXAFS spectra of the intermediate phase is lower than for the structural data, which can be explained

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**Figure 2.** Kinetics of the precipitation of Pu starting from Pu$^{VI}$ at pH 11 ([Pu]$=6\times10^{-5}\text{ m}$). Inset: Crystal structure of the formed phases.

**Figure 3.** Pu L$_3$ edge HERFD spectra of PuO$_2$, and the Pu$^{IV}$ intermediate phase formed during the synthesis of PuO$_2$ nanoparticles from Pu$^{VI}$ precursors at pH 11. Experimental data (black lines) are compared with FDMNES calculations for bulk PuO$_2$ and NH$_4$PuO$_2$CO$_3$ (red lines).
either by a (partially) amorphous nature or by nano-sized particles.

The intermediate NH₄PuO₂CO₃ phase was completely dissolved within \( \approx 10 \) h, after which the PuO₂ NPs were formed as a result of longer redox reactions (see Figure 2). Finally, a part of the intermediate Pu\(^V\) phase was centrifuged out of suspension and dried at room temperature in order to check its stability over months. Surprisingly, the dried NH₄PuO₂CO₃ phase was found to be stable over months. We recorded additional Pu L₃ HERFD spectra after 3 months and the spectral shape remained the same (Figure S8). Therefore, the method reported here can be used to synthesize this Pu\(^V\) phase.

To understand the pH influence, we performed a similar experiment at pH 8. The kinetics of the Pu precipitation is very similar to the experiment at pH 11, whereas the quantity of the intermediate Pu\(^V\) phase is lower (Figure S9). Comparison of the experimental conditions with the available thermodynamic data shows that Eh/pH values during our synthesis correspond to the area of stability of the Pu\(^V\) phase close to the phase boundary (Figure S10). It makes the formation of the intermediate Pu\(^V\) phase possible, but at the same time, the high thermodynamic stability of PuO₂ and its extremely low solubility lead to a further transformation of the Pu\(^V\) phase into PuO₂.

We show here for the first time that while Pu\(^V\) solid-state complexes are always viewed as exotic compounds, a thermodynamically metastable Pu\(^V\) solid phase is formed during the reductive precipitation of PuO₂ NPs from a Pu\(^V\) precursor at pH 11. The intermediate Pu\(^V\) phase is characterized for the first time using HERFD at the Pu M₁ edge and model calculations in the framework of AIM. The Pu M₁ HERFD method allows for the unambiguous identification of the Pu oxidation state, it demonstrates the Pu\(^V\) existence, and provides quantitative estimates for varying Pu oxidation states. The local structure of the intermediate Pu\(^V\) phase, similar to NH₄PuO₂CO₃, is identified by a combination of the Pu L₃ HERFD experiment and ab-initio calculations, and is found to be stable over a period of several months. The redox reactions behind aqueous Pu\(^V\)–PuO₂ NPs and the formation of Pu\(^V\) cause the substantial increase of the solubility. This finding provides a significant step towards a better understanding of Pu chemistry and emphasizes the value of the HERFD technique for studies of PuO₂ NPs formation under different conditions.

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Conflict of interest

The authors declare no conflict of interest.

Keywords: actinide chemistry · electronic-structure calculations · pentavalent plutonium · plutonium dioxide nanoparticles · Pu M₁ HERFD

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