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Perfect isolation of $\pi$-conjugated molecules on inorganic surfaces with [1]rotaxane structure for enhancing electrical properties

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Contributions

H.M. and J.T. designed and directed the project. S.-Y. C. performed the synthetic experiments and collected the central data. O.M. collaborated with S.-Y. C. to investigate the interfacial charge transfer. G.S. and Y.Y. performed and interpreted the electrocatalysis. Y.K. and H.T. synthesized the Co-catalyst. T.Ka. and H.O. manufactured the single-crystalline ITO substrates. T.Ko., A.K. and R.A. investigated the optical analyses of the modified ITO substrates. T.T. and K.U. performed and analyzed XPS. S.-Y. C. and H.M. wrote the manuscript. All authors have approved the manuscript.

Abstract

π-Conjugated molecules have been utilized to functionalize inorganic surfaces to form organic–inorganic hybrid materials. However, the intrinsically strong π–π interaction results in undesirable aggregations on the inorganic surface, thereby disturbing the charge transfer through the organic–inorganic interface. In this
study, a new strategy was developed using insulated π-conjugated molecules bearing a [1]rotaxane structure, where the π-conjugation was covered with covalently linked permethylated α-cyclodextrins. Aggregation-free immobilization was achieved on an inorganic surface by using insulated molecules to suppress intermolecular interaction. In the presence of these insulated molecules, the hybrid interface displayed excellent interfacial electrical properties. Moreover, the functionalized hybrid surface was utilized as an electrocatalyst to produce hydrogen peroxide using a Co(II)–chlorin complex, wherein the catalytic efficiency was improved dramatically by utilizing insulated molecules as bridging moieties at the interface. These results demonstrate that the insulation of π-conjugated molecules is a powerful strategy for modifying inorganic surfaces.

Introduction

Immobilization of organic molecules onto inorganic surfaces synergistically integrates the features of both components, resulting in exceptional material properties. In particular, π-conjugated molecules, which possess unique optical and electrical properties, have been utilized for diverse organic–inorganic hybrid devices such as memory devices, solar cells, and biosensors. Particularly, in the case of electrical devices, the high molecular modification density and high charge-transfer efficiency at the interface, which are governed by the morphology of the surface organic components, are critical to the device performance. Conventional surface modifications of inorganic materials using π-conjugated molecules have been achieved via self-assembled monolayers (SAMs), covalent bond formations, and depositions. However, these classical methods are only suitable for simple and limited conjugated molecules (i.e., unexpanded π-conjugation) to provide uniform and ordered hybrid interfaces. Although π-expanded conjugated molecules offer high functionalities for such electrical devices, they readily form objectionable aggregation on surfaces, owing to their strong π–π interaction (Fig. 1a). This aggregation induces disordering and large protrusions on the surface; therefore, rather than direct injection into the inorganic materials, unanticipated charge transfer between the adjacent molecules occurs on the surface of these materials. This decreases the charge-transfer efficiency of the material. Thus, π-expanded conjugated
molecules for electrical devices result in undesirable aggregation, limiting the improvement of the functional device performance. Paving the optimum conductive pathways at the interface by inhibiting π-aggregation is therefore critical to improve the material performance, especially for electrical devices with π-expanded conjugated molecules.

To suppress π-aggregation, an isolated environment is typically created around π-expanded conjugated molecules on the material surface by utilizing aliphatic compounds. Mixed SAM methods, where the conjugated molecules are diluted in a non-conjugated molecular monolayer, provide an isolated environment around the π-conjugation. However, such methods require a low π-conjugation composition ratio, which is disadvantageous for electric devices. On the other hand, increasing the concentration of π-conjugated molecules induces the formation of aggregation structures on the surface of the material. Similarly, some π-conjugated systems bearing bulky side chains to suppress π–π interaction afford uniform surface monolayers. However, such ordered systems are attributed to the appropriate surface molecular interaction, and further π-expansion would also result in their aggregation. Overall, except for low-concentration immobilization, most π-expanded aromatic compounds cannot be immobilized on surfaces without aggregation. Accordingly, high-density uniform immobilization of π-expanded functional conjugation at the hybrid interface remains a challenging target.

![Fig. 1 Conceptual illustration of modification of inorganic surface using π-conjugated molecules. a SAMs constructed by unexpanded π-conjugated molecules (top) and intermolecular charge transfer caused](image-url)
by undesirable aggregation due to strong aromatic interaction derived from the expansion of the \( \pi \)-conjugated backbone (bottom). b Non-aggregation and efficient charge injection because of \( \pi-\pi \) interaction and intermolecular charge-transfer inhibition using [1]rotaxane structure. c Molecular design of the conjugated molecules bearing the [1]rotaxane structure for surface immobilization.

Recently, rotaxane structures have gained importance as a promising method to inhibit \( \pi-\pi \) interaction.\(^{21,22}\) The cyclic molecules efficiently cover the \( \pi \)-conjugated axles and inhibit unfavorable \( \pi \)-aggregation.\(^{23,24}\) In particular, pseudorotaxanes, in which cyclic and axis molecules dynamically form threading structures,\(^{25,26}\) have been utilized for hybrid modification because of their facile in situ preparation.\(^{27}\) However, owing to their lack of stoppers, pseudorotaxanes can dynamically dissociate to form irregular structures on the hybrid interfaces, including aggregated \( \pi \)-conjugated molecules and empty cyclic molecules. These unclear and aggregated structures decrease the electrical properties of the hybrid interface. Consequently, examples of hybrid materials utilizing pseudorotaxanes to improve the electrical performance in terms of effective electron injections at the hybrid interface have rarely been reported.\(^{28-31}\) Therefore, a new strategy for highly advanced molecular designing of \( \pi \)-expanded conjugated molecules, to attain fine interface immobilization, would create new platforms toward further growth in this field of nanoscience.

To address these concerns, we introduced the [1]rotaxane strategy (Figs. 1b and 1c).\(^{32-36}\) In contrast to common rotaxane structures, [1]rotaxane structures prevent macrocycles from dissociating, without any bulky stopper units in the rotaxanes. The defect-free insulation improves their conversion ratio on the axles and ensures that the conjugated molecules are highly insulated. As a result, the physical properties derived from their conjugation are preserved from the effect of their \( \pi-\pi \) interaction, even in the solid state, because of the high proportion of covering structures.\(^{37-41}\) Their advantage as highly protected structures inspired the design of a novel insulated conjugated molecule for immobilization on inorganic surfaces. The defect-free structures contribute non-aggregation structures with high density and uniform \( \pi \)-expanded conjugation immobilization at the hybrid interface to inhibit intermolecular charge transfer. Moreover, the finely
controlled interface at the molecular scale provides high-performance electrical devices via efficient charge
transfer through each π-conjugated backbone. In this study, we synthesized an insulated conjugated
molecule to achieve independent immobilization on inorganic substrates. The molecule comprises
phenylene-ethynylene (PE)-based π-conjugation covered by a linked permethylated α-cyclodextrin (PM α–
CD) as the shielding unit, and displays high solubility in organic solvents together with a deep cavity (Fig.
1c). As an anchoring moiety to the substrates, phosphonic acid was directly introduced at the end of the
conjugated section to allow strong interaction and to transfer the charge to various metal oxide substrates\textsuperscript{9,42}
such as indium tin oxide (ITO) and fluorine-doped tin oxide (FTO), which are utilized as electrodes in
electrical devices. The insulation effects were investigated to clarify the high-degree independency and
excellent electrical properties on the hybrid interface compared to those on the uninsulated counterparts,
owing to the inhibition of intermolecular interactions by three-dimensional encapsulation. In addition, the
hybrid system based on [1]rotaxanes was applied to electrocatalysis, revealing marked improvement in the
catalytic efficiency. The results demonstrated that enhanced charge-transfer and catalytic efficiencies were
achieved because the [1]rotaxane strategy efficiently isolated the π-conjugated molecules to inhibit π–π
interaction. This strategy shows great potential for application to various electric devices.

Results

Synthesis of Insulated Junction Molecules

The immobilization behavior of cyclic-insulated molecules was investigated using insulated
conjugated molecule 3. Precursor 1 bearing PM α–CD and terminal ethyl-protected phosphonic acid was
prepared according to previously reported procedures\textsuperscript{43}. Moreover, 1 was converted into the corresponding
insulated structure 2 quantitatively by hydrophobic–hydrophilic interactions in a heated (60 °C) high-
polarity solvent (MeOH/H\textsubscript{2}O = 1:1). This was followed by deprotection of the ethyl ether groups in 1 and
2 using trimethylsilyl bromide (TMSBr) and triethylamine (TEA) in dichloromethane (CH\textsubscript{2}Cl\textsubscript{2}) to obtain
the uninsulated and insulated phosphonic acid-derived molecules $3'$ and $3$, respectively, (Fig. 2a). The high activation barrier for the threading/dethreading transformations$^{44}$ was confirmed in previous works$^{45-48}$ and hence, heating the solvent mixture was essential for converting $1$ to $2$ and vice versa. Thus, these structures displayed kinetic stability at ambient temperature. The high kinetic stability at room temperature allowed the selective preparation of $3'$ and $3$ via deprotection of the ethyl ether groups in $1$ and $2$, respectively. In the $^1$H NMR spectra (Fig. 2b), the chemical shifts in the aromatic region of $3$ were clearly shifted downfield relative to those of the uninsulated counterpart $3'$, owing to deshielding by the threading structure.$^{46}$ The loss of ethyl proton signals in the aliphatic region confirmed successful deprotection without any unwanted changes to the insulated/uninsulated structures. In addition, insulated molecule $3$ was kinetically stable even in low-polarity solvents, in which the uninsulated structures are thermodynamically favored. Indeed, the insulation remained intact after storage in CHCl$_3$ at room temperature for one day.

**Insulation Effects on the Surface Morphology**

The insulation effects on the molecular morphologies of $3'$ and $3$ after their immobilization on the metal oxide surface were next investigated. To establish whether the surface protrusions were created by the immobilized species or the surface substrate itself, commercially unavailable single-crystalline ITO substrates with ultra-flat surfaces (root mean square roughness, <0.2 nm), fabricated by pulsed laser deposition on single crystal YSZ plate, were utilized.$^{49,50}$ Compared to those of amorphous ITO, the individual heights of $3'$ and $3$ on the single-crystalline ITO surfaces were easily and correctly analyzed in detail using atomic force microscopy (AFM). The crystalline ITO substrates were modified by dipping into a low-concentration (50 μM) MeOH solution of $3'$ or $3$. The low concentration prevented high-density adsorption of the conjugated molecules on the ITO surface, thereby enabling the analysis of the individual height of each adsorption species by AFM. Consequently, the AFM image revealed that the single-crystalline ITO substrate, which showed a regular step-and-terrace structure, similar to that reported by Ohta et al.,$^{49,50}$ was modified by the numerous protrusions of insulated conjugated molecule $3$ (Fig. 2c). The height distribution of this molecule on the single-crystalline ITO surface mostly ranged from 0.9 to 1.9 nm$^{51}$
(Fig. 2e), which approached the calculated length of 3 (1.7 nm; Fig. S2). This indicated that the observed structure was the attached single molecule. The formation of huge protrusions, owing to undesirable aggregation of the uninsulated conjugated molecule 3' on the single-crystalline ITO surface, was observed (Figs. 2d and 2f). In the histogram of immobilized 3' (Fig. 2f, inset), the height frequency exceeded 2 nm, which was considered as aggregation, in more than 10% of the graph. This indicated that for uninsulated 3', surface aggregation occurred even when a low-concentration solution was utilized for surface modification. Indeed, although uninsulated 3' possessed PM α–CD as a bulky side chain, the strong intermolecular interactions between the expanded conjugation resulted in an unfavorable aggregated structure on the ITO substrate.

Fig. 2 Synthesis of insulated/uninsulated molecules and morphology of the modified single-crystalline ITO substrates. a Synthetic route of insulated/uninsulated molecules (3/3'). b Aromatic region in the $^1$H
NMR spectra of (top) 3' and (bottom) 3. AFM images of the single-crystalline ITO substrates modified by e 3 and d 3' with the plan views and cross-sectional profiles. Height of each protrusion of e immobilized 3 (99 counts; inset: histogram of height analyses) and f immobilized 3' (70 counts; inset: histogram of height analyses) in the AFM image.

Insulation Effects on the Interface Electronic Properties

To analyze the electrical properties of the immobilized molecules, 4 and 4' bearing redox-active ferrocene units were next prepared (Scheme S3). The surface coverage concentration (\( \Gamma \)) and other electrical properties were then determined using cyclic voltammetry. The experiment for calculating the \( \Gamma \) values was carried out in a 0.1 M solution of tetrabutylammonium hexafluorophosphate (TBAPF\(_6\)) in CH\(_2\)Cl\(_2\). \( \Gamma \) values were then obtained from the cyclic voltammograms using the formula:

\[
\Gamma = \frac{Q}{nFA}
\]

where \( Q \) is the charge obtained by integrating the anodic peak in the cyclic voltammogram (gray area in Fig. 3a), \( n \) is the number of electrons transferred during the redox process (\( n = 1 \) for 4 and 4'), \( F \) is the Faraday constant (\( F = 9.65 \times 10^4 \) s A mol\(^{-1}\)), and \( A \) is the immersed area of the electrode. The maximum \( \Gamma \) value of 4 was observed when the ITO substrate was immersed into a 1000 \( \mu \)M MeOH solution of 4 (Table S2). Subsequently, the ITO substrate was also immersed into a 1000 \( \mu \)M MeOH solution of 4'. The \( \Gamma \) values of insulated 4 and uninsulated 4' are summarized in Table S3. The \( \Gamma \) value of 4' was approximately twice that of 4 because of the immense protrusions observed on the AFM images, which were created by aggregation of uninsulated 4' on ITO (Fig. 3c). The cyclic voltammograms of the 4-modified ITO electrode were next recorded at different scan rates. The plot of peak current versus scan rate (Figs. 3b and S3a) revealed a linear relationship, confirming that 4 was adsorbed on the electrode surface. The peak current corresponded to the current from the electrode-adsorbed redox species according to the formula:

\[
i_p = \frac{n^2F^2}{4RT} vA\Gamma
\]
where $i_p$ is the peak current and $v$ is the scan rate. In addition, there was negligible change in the $\Gamma$ value of immobilized 4 when the potential sweep was repeated 25 times (Figs. S4a and S4b). This indicated that the modified surface prepared utilizing the insulated molecule 4 displayed significant redox durability.

In the cyclic voltammograms, the separation between the anodic and cathodic peaks of the redox-active species immobilized on the electrode surface is designated as 0 mV under ideal reversible conditions. At a scan rate of 100 mV s$^{-1}$, the peak-to-peak separation was 3 mV for both immobilized 4 (Fig. 3a) and immobilized 4' (Fig. S3b). This value is small enough to be in excellent agreement with the ideal value for the reversible response of a surface-adsorbed species. Moreover, the value of the full width at half maximum (FWHM) in the cyclic voltammograms was utilized to assess the immobilized molecules at the point of the electrostatic interaction with the neighboring species, i.e., as the repulsion forces were dominant, the redox peak was wider than the ideal width (90.6/n mV at 25 °C, $n = 1$ for 4 and 4'), while when the attraction forces were dominant, the redox peak became narrower than the ideal width. Moreover, while the FWHM value of immobilized 4' (77 mV) was smaller than the ideal value (Fig. S3b), that of immobilized 4 (88 mV) approached the ideal value (Fig. 3a). The smaller FWHM of 4' on the ITO surface was attributed to the electrostatic interaction between neighboring molecules due to unwanted aggregation. On the other hand, the intermolecular interaction between the molecules of 4 on the ITO surface was efficiently inhibited by the [1]rotaxane structure, providing isolated π-conjugated cores even at the organic–inorganic interface.
Fig. 3 Interfacial electronic properties and morphology of modified ITO substrates. a Cyclic voltammogram of immobilized 4 at a scan rate of 100 mV s\(^{-1}\). Conditions: 100 mM CH\(_2\)Cl\(_2\) solution of TBAPF\(_6\). b Plot of anodic and cathodic peak currents of immobilized 4 versus the scan rates. c AFM images of the ITO substrate modified with (left) 4 and (right) 4\(\prime\).

The increase in peak-to-peak separation with faster scan rates was attributed to the slow charge transfer between the ITO and immobilized redox-active species.\(^{57,58}\) To investigate the insulation effects on the charge transfer between the ferrocene unit and ITO electrode, the variation in the peak-to-peak separation potentials (Fig. 4a; \(\Delta E = E_p - E^\circ \prime\), where \(E_p\) is the peak potential and \(E^\circ \prime\) is the formal potential) with increasing scan rate was determined to construct the trumpet plots (Fig. 4b).\(^{57,59,60}\) While the peak-to-peak separation of uninsulated 4\(\prime\) occurred at \(\log(\nu) = 0.6\) (\(\nu = 4\) V s\(^{-1}\)), insulated 4 did not display any significant peak-to-peak separation, even at a high scan rate of \(\log(\nu) = 2\) (\(\nu = 100\) V s\(^{-1}\)). This indicated that the rate of charge transfer between the ITO electrode and 4 increased, owing to the insulated structure.
In our previous study on the single molecular conductance of insulated molecules, the conductance of single insulated conjugation was lower than that of the corresponding uninsulated molecule. This occurred because the π-conjugated backbone was twisted, owing to insulation by PM α–CD.\textsuperscript{61} In this study, however, the insulated structure improved the charge-transfer efficiency relative to the uninsulated counterpart, indicating that the insulation enhanced the charge transfer at the interface rather than on the π-conjugated core. The highly efficient charge transfer between 4 and the ITO electrode were attributed to the inhibition of π–π interaction and intermolecular charge transfer. Indeed, the intermolecular charge transfer between the bare π-conjugated cores of uninsulated 4' decreased the charge-transfer efficiency. Thus, appropriate π-aggregation inhibition at the organic–inorganic interface utilizing [1]rotaxanes structures is a promising strategy for developing exceptional electrical devices, owing to the high charge-transfer efficiency.

**Fig. 4 Interfacial charge transfer.** a Cyclic voltammograms of immobilized 4 and 4' at various scan rates. b Trumpet plot for immobilized 4 and 4'. Conditions: 0.1 M CH\textsubscript{2}Cl\textsubscript{2} solution of TBAPF\textsubscript{6}. 
Surface Functionalization with Metalloporphyrin

The isolated surface π-conjugations were effectively applied to electrical devices based on organic–inorganic hybrid materials. To provide functionalities on the hybrid materials, we focused on the immobilization of metal complexes through metal–ligand coordination, which have been widely utilized for dyes and surface inorganic catalysis.\textsuperscript{62,63} In particular, the modification of metalloporphyrin analogs onto metal oxide surfaces have been effectively applied to artificial photosynthesis and electrosynthesis devices.\textsuperscript{64,65} In the current study, the 4-pyridyl group was introduced as a coordination point into insulated and uninsulated structures to obtain 5 and 5', respectively (Scheme S4). To evaluate the potential of 5 for application to surface engineering, we first investigated the complexation behavior between 5 (or 5') and \textit{Rh}^{III}\textit{Cl}(OEP) (OEP: octaethylporphyrin), both in the solution system and on the surface system. In the former, the coordination between compound S12\textsuperscript{66} and \textit{Rh}^{III}\textit{Cl}(OEP) was confirmed from the characteristic up-field shifts observed for complexation in the $^1$H NMR spectra (Figs. S5 and S6). In addition, the complexation was applied to the surface modifications. Thus, the ITO substrate was immersed in a solution of 5 or 5' in MeOH (1000 μM) overnight, and subsequently submerged in a solution of \textit{Rh}^{III}\textit{Cl}(OEP) in CH$_2$Cl$_2$ (200 μM) for 2 h to obtain \textit{Rh}^{III}\textit{Cl}(OEP)–5/ITO (Fig. 5a) or \textit{Rh}^{III}\textit{Cl}(OEP)–5'/ITO, respectively. In the AFM images of the modified substrates (Fig. S7), there were no distinct aggregation structures on the surface of \textit{Rh}^{III}\textit{Cl}(OEP)–5/ITO. In contrast, many large protrusions were observed as aggregation structures on its uninsulated counterpart. Surface modification of the Rh–porphyrin complex was confirmed by UV–vis absorption and XPS analysis. In the differential UV–vis absorbance spectrum of \textit{Rh}^{III}\textit{Cl}(OEP)–5/ITO (Fig. S8b, red line), the characteristic Soret and Q bands of porphyrin were observed. Conversely, no characteristic porphyrin absorption bands were observed following the direct immersion of the ITO substrate into the \textit{Rh}^{III}\textit{Cl}(OEP) solution without immersion into a solution of 5 (Fig. S8b, blue line). Moreover, the maximum absorption wavelength of \textit{Rh}^{III}\textit{Cl}(OEP)–5/ITO corresponded to that observed in the \textit{Rh}^{III}\textit{Cl}(OEP)–S12 solution (Fig. 5b). The XPS survey spectra of \textit{Rh}^{III}\textit{Cl}(OEP)–5/ITO also showed Rh 3d peaks at 310 and 315 eV (Fig. 5c). These results strongly support...
the formation of \textit{Rh}^{III}\textit{Cl(OEP)}–5 complexes with ITO as the hybrid interface. According to the integral areas of the Rh 3d\textsubscript{3/2} and P 2p peaks, which were normalized by that of the In 3d peak derived from the ITO substrate, the ratio of Rh-porphyrin-to-insulated 5 in \textit{Rh}^{III}\textit{Cl(OEP)}–5/ITO was 1:1 (Table S4). On the other hand, that between Rh-porphyrin and 5' in \textit{Rh}^{III}\textit{Cl(OEP)}–5'/ITO was 1.7:1, because of the non-independence of the uninsulated molecule on the surface. According to the space filling model obtained by calculation (Fig. S10), the molecular size of \textit{Rh}^{III}\textit{Cl(OEP)} was near-identical to the diameter of PM α–CD. This permitted high-density packing of the metalloporphyrin portion, while simultaneously avoiding metalloporphyrin overlapping on the metal oxide surface. Thus, this sequentially coordinated modification utilizing insulated 5 allowed an easy approach for introducing a large metalloporphyrin of high crystallinity and low solubility onto the metal oxide surface via metal–ligand axial coordination. This was achieved without complicated synthesis of the metalloporphyrin, and bypassed aggregation structure formation on the metal oxide surface.

\textbf{Fig. 5 Coordination of metalloporphyrin and its analog to insulated molecules.} a Image of \textit{Rh}^{III}\textit{Cl(OEP)}–5/ITO and \textit{Co}^{II}(Ch)–5/FTO. b Normalized UV–vis absorption spectra of the \textit{Rh}^{III}\textit{Cl(OEP)}–5/ITO and CHCl\textsubscript{3} solution of the \textit{Rh}^{III}\textit{Cl(OEP)}–S12 \((1\times10^{-5} \text{ M})\) complex. c XPS spectra...
of Rh 3d$_{3/2}$ (green line) and Rh 3d$_{5/2}$ (yellow line) for Rh$_{III}$Cl(OEP)-5/ITO substrate. d H$_2$O$_2$ production of Co$^{II}$(Ch)-5/FTO (solid line), Co$^{II}$(Ch)-5'/FTO (dashed line), and Co$^{II}$(Ch)/FTO (dotted line) as working electrodes. The reactions were carried out by applying constant voltage ($E = 0.41$ V vs. Ag/AgCl) in an O$_2$-saturated perchlorate buffer solution (pH 1.3, 100 mM, 11 mL).

Electrocatalysis for H$_2$O$_2$ generation

Surface engineering using insulated molecules and metalloporphyrin derivatives was next applied to the electrical devices. Cobalt chlorin complexes [Co$^{II}$(Ch)] have been reported as highly efficient and selective molecular catalysts for the two-electron selective reduction of dioxygen (O$_2$) to yield hydrogen peroxide (H$_2$O$_2$) in homogeneous solution. However, the selectivity decreases when the catalysts are immobilized on the electrode surface. Owing to the disordered morphology of the surface-adsorbed Co$^{II}$(Ch), a $\mu$-1,2-peroxo dinuclear structure is formed, which carries on the four-electron-reduced dioxygen, producing H$_2$O as a by-product. In this study, the high-degree independency and outstanding electrical properties of a [1]rotaxane-based hybrid system were applied to the Co$^{II}$(Ch)-catalyzed electrochemical device, to prevent the formation of $\mu$-1,2-peroxo dinuclear structures and enhance the charge-transfer efficiency. Insulated 5 and uninsulated 5' coordinated to Co$^{II}$(Ch) were immobilized on FTO to form Co$^{II}$(Ch)-5/FTO$^7$ (Fig. 5a) and Co$^{II}$(Ch)-5'/FTO electrodes through the Langmuir–Blodgett (LB) technique at the same surface pressure, respectively. Both metal complexes formed a solid condensed monolayer on the FTO surface according to the pressure–area ($\pi$–$A$) isotherms (Fig. S12). In addition, the Soret and Q bands of Co$^{II}$(Ch) were observed in the differential absorption spectra of modified FTO in both Co$^{II}$(Ch)-5/FTO and Co$^{II}$(Ch)-5'/FTO (Fig. S13).

During the electrocatalytic synthesis, the Co$^{II}$(Ch)-5/FTO system showed the best H$_2$O$_2$ production efficiency (Fig. 5d). In contrast, the Co$^{II}$(Ch)/FTO system, which directly deposited the catalyst on the FTO surface by the LB technique, exhibited the lowest production efficiency. The H$_2$O$_2$ production of the Co$^{II}$(Ch)/FTO system declined after 1 h because of catalyst desorption, owing to the structure without
phosphonic acid-based-anchoring portion. In addition, the catalyst films formed a \( \mu \)-1,2-peroxo dinuclear structure on the surface which, as previously mentioned, yielded \( \text{H}_2\text{O} \) as a byproduct, thereby decreasing the selectivity. Unlike the \( \text{Co}^{II}(\text{Ch})/\text{FTO} \) system, the \( \text{Co}^{II}(\text{Ch})-5'/\text{FTO} \) system contained an anchoring portion to improve the adsorption stability. Although \( \text{Co}^{II}(\text{Ch})-5'/\text{FTO} \) exhibited a similar adsorption density to that of its insulated counterpart in line with the absorption spectra (Fig. S14), the performance of the \( \text{Co}^{II}(\text{Ch})-5'/\text{FTO} \) system was inferior. According to the previously described charge-transfer experiment, the uninsulated structure of 5' was detrimental to charge transfer. This was suggested to be one of the reasons for the lower production efficiency compared to that of the \( \text{Co}^{II}(\text{Ch})-5/\text{FTO} \) system. In addition, because of the dependence on the uninsulated structure, the uninsulated \( \text{Co}^{II}(\text{Ch})-5' \) complex formed a monolayer with intrinsically loose alignment on the FTO surface, while part of the \( \text{Co}^{II}(\text{Ch}) \) portion formed a \( \mu \)-1,2-peroxo dinuclear structure, decreasing the selectivity. Consequently, improvement in the production efficiency of the \( \text{Co}^{II}(\text{Ch})-5/\text{FTO} \) system should be considered independently of the insulated surface structure and charge transfer improvement.

**Discussion**

In summary, \([1]\)rotaxane molecules bearing a phosphonic acid-derived PE as the conjugated backbone and PM \( \alpha \)-CD as a protective macrocycle were immobilized on metal oxide surfaces via a wet process. AFM and CV analyses revealed the insulation effects on the hybrid system of the \([1]\)rotaxanes. The insulated molecules were immobilized on the metal oxide surfaces in ideal state without aggregation and displayed high charge-transfer efficiency at the interface, as compared to their uninsulated counterparts. The high advantages of insulation were applied to electrocatalysis as electrical hybrid devices. The \([1]\)rotaxane system was utilized for the platform to introduce organic functionalities on the inorganic electrode. The catalytic efficiency of the cobalt chlorin complex was markedly improved by utilizing insulated molecules. These results indicate the importance of the \([1]\)rotaxane strategy in isolating molecules from unfavorable molecular interactions, even at the hybrid interface, providing excellent performance in electrical hybrid devices. This methodology possesses high potential to upgrade the performances of
existing electrical devices based on π-conjugated hybrid systems to outstanding devices, by improving the
independency and charge-transfer efficiency of the π-conjugated molecules at the hybrid interface. The
[1]rotaxane strategy can be considered as a general and versatile method for interfacial control, even for
other types of hybrid junctions in addition to phosphonic acid and tin oxides, which would efficiently solve
the conventional problems observed in hybrid interfaces composed of π-conjugated molecules and
inorganic materials.

**Methods**

**Synthesis of 1**

1,4-diiodobenzene (2.4 g, 7.27 mmol) was dissolved in degassed iPr₂NH/THF (6/3 mL). Under an argon,
S1 (1.00 g, 737 μmol), PdCl₂(PPh₃)₂ (51.8 mg, 73.8 μmol) and CuI (7.0 mg, 36.8 μmol) were added to the
solution. The reaction mixture was stirred under an argon at room temperature overnight. The solvent was
removed *in vacuo*, and the residue was purified by column chromatography on silica gel (1:1 toluene:EtOAc
and 85:15 EtOAc:MeOH) to yield S2 as an orange solid (845 mg, 542 μmol, 74%).

Under an argon atmosphere, **Pd(OAc)₂** (11.6 mg, 51.6 μmol), dppf (57.3 mg, 103 μmol) and KOAc (50.6
mg, 516 μmol) were placed in the reaction vessel. THF (20 mL) was introduced and the mixture was stirred
and heated at 68 °C. After 5 min, S2 (805 mg, 517 μmol), diethyl phosphite (666 μL, 5.16 mmol) and
triethylamine (TEA) (357 μL, 2.58 μmol) were added to the mixture. The reaction mixture was stirred at
68 °C overnight. The mixture was dried *in vacuo*, and the residue was purified by GPC with CHCl₃ as the
eluent to yield 1 as an orange solid (609 mg, 388 μmol, 72%). *ESI MS: (m/z) 1568.665 ([M+H]+, C₇₁H₁₁₁NO₃₅P, calcd. 1568.667).*¹H NMR (500 MHz, CDCl₃, r.t.), δ: 7.82–7.80 (m, 4H, ArH), 7.71–7.69
(m, 2H, ArH), 7.61 (d, J = 8.5 Hz, 1H, ArH), 5.09–2.99 (m, 97H, CD−H, OCH₃, OC₂H₃), 1.32 (t, J =
6.3 Hz, 6H, OCH₂CH₃).¹³C NMR (126 MHz, CDCl₃, r.t.), δ: 159.55, 148.20, 133.46, 131.72 (d, J = 12.6 Hz),
131.72 (d, J = 12.6 Hz), 129.08 (d, J = 189.8 Hz), 126.50, 126.47, 119.4, 116.0, 106.7, 100.44, 100.29,
Synthesis of 2

1 (30 mg, 19 μmol) was added into MeOH/H₂O (1/1, 30 mL), and the solution was stirred at 60 °C overnight. The reaction mixture was evaporated to reduce MeOH. The mixture was diluted with CHCl₃ and washed with water. The organic layer was separated and dried over MgSO₄, and then filtered. The solvent was removed by evaporation to yield 2 as a pale yellow solid (27 mg, 17 μmol, 89%) without further purification.

ESI MS: (m/z) 1590.648 ([M+Na]+, C₇₁H₁₁₀NO₃₅PNa, calcd. 1590.649). ¹H NMR (500 MHz, CDCl₃, r.t.), δ: 8.23−8.22 (m, 2H, ArH), 8.00−7.98 (m, 4H, ArH), 7.80 (d, J = 7.9 Hz, 1H, ArH), 5.06−2.99 (m, 97H, CD−H, OCH₃, OC₃H₂CH₃), 1.26 (dd, J = 11.3, 6.7 Hz, 6H, OCH₂CH₃). ¹³C NMR (126 MHz, CDCl₃, r.t.), δ: 162.06, 148.17, 134.05, 132.32 (d, J = 13.9 Hz), 131.41 (d, J = 10.1 Hz), 130.35 (d, J = 191.5 Hz), 125.19, 123.28, 117.90, 116.81, 100.72, 100.54, 100.11, 99.98, 99.90, 97.96, 97.19, 87.39, 83.56, 83.26, 82.90, 82.54, 82.37, 82.34, 82.28 (peaks overlapped), 82.02 (peaks overlapped), 81.93, 81.91, 81.65, 81.34, 81.15, 81.09, 81.03, 80.98, 72.03 (peaks overlapped), 71.92, 71.57, 71.48, 71.44, 71.33 (peaks overlapped), 71.26, 71.12, 70.62, 70.48, 61.96, 61.93, 61.84, 61.79, 61.71, 61.66, 61.58, 61.56, 58.99, 58.88, 58.81, 58.57, 58.51, 58.30, 58.01, 57.70, 57.62, 57.54, 57.46, 16.12 (d, J = 6.3 Hz). ³¹P NMR (202 MHz, CDCl₃, r.t.), δ: 16.46.

Synthesis of 3'

Under a nitrogen, 1 (30.5 mg, 19.4 μmol) was dissolved in TEA/CH₂Cl₂ (0.16/2 mL), and TMSBr (49 μL, 383 μmol) was added into the solution. After the reaction mixture was stirred at room temperature for 12 h, the solvent was removed by vacuum distillation and 5 mL of methanol was added to the resulting crude.
After the dissolved crude was stirred at room temperature for another 12 h, the solvent was evaporated in vacuo. The residue was dissolved in CHCl₃ and washed by dilute aqueous HCl solution. The organic layer was separated and dried over MgSO₄, then the solvent was removed in vacuo. Furthermore, the residue was dissolved in MeOH and washed by hexane. The MeOH layer was collected and concentrated to yield 3' as a pale yellow solid (28.0 mg, 18.5 μmol, 95%). Surface immobilizations were carried out without further purifications. 

ESI MS: \( m/z \) 1510.591 ([M–H]⁻, \( C_{67}H_{101}NO_{35}P \), calcd. 1510.590). 

\(^1H\) NMR (500 MHz, CDCl₃, r.t.), δ: 7.83–7.77 (m, 4H, ArH), 7.66–7.62 (broad, 2H, ArH), 7.56 (d, \( J = 8.1 \) Hz, 1H, ArH), 5.07–2.97 (m, 93H, CD–H, OCH₃). 

\(^13C\) NMR (126 MHz, CDCl₃, r.t.), δ: 159.54, 148.10, 133.21, 131.50 (d, \( J = 16.4 \) Hz), 131.00 (d, \( J = 10.1 \) Hz), 125.98, 119.43, 115.90, 106.93, 100.29 (peaks overlapped), 100.17 (peaks overlapped), 100.09 (peaks overlapped), 99.98 (peaks overlapped), 97.48, 86.53, 82.63, 82.55, 82.46 (peaks overlapped), 82.38, 82.16 (peaks overlapped), 82.10 (peaks overlapped), 82.02, 81.96, 81.76, 81.25, 81.17 (peaks overlapped), 81.11, 71.69, 71.66, 71.49 (peaks overlapped), 71.28 (peaks overlapped), 71.23, 71.17, 70.34, 68.30, 67.94, 61.86, 61.79 (peaks overlapped), 61.74 (peaks overlapped), 61.72, 59.30, 59.09, 59.08, 59.06, 58.87, 58.16, 57.87, 57.81 (peaks overlapped), 57.77, 57.34. 

\(^31P\) NMR (202 MHz, CDCl₃, r.t.), δ: 18.74.

Synthesis of 3

Under a nitrogen, 2 (27.6 mg, 17.6 μmol) was dissolved in TEA/CH₂Cl₂ (0.15/2 mL), and TMSBr (45 μL, 383 μmol) was added into the solution, and the reaction mixture was stirred at room temperature for 12 h. After the solvent was removed by vacuum distillation, 5 mL of methanol was added to the resulting crude and stirred at room temperature for another 12 h. Then, the solvent was evaporated by vacuum. After the residue was dissolved in CHCl₃ and washed by dilute aqueous HCl solution, the organic layer was separated and dried over MgSO₄ and the solvent was removed in vacuo. Furthermore, the residue was dissolved in MeOH and washed by hexane. The MeOH layer was separated and concentrated to yield 3 as a pale yellow solid (16.4 mg, 10.8 μmol, 61%). Surface immobilizations were carried out without further purifications.

ESI MS: \( m/z \) 1510.593 ([M–H]⁻, \( C_{67}H_{101}NO_{35}P \), calcd. 1510.590). 

\(^1H\) NMR (500 MHz, CDCl₃, r.t.), δ: 8.22
Synthesis of 4’

Under a nitrogen, S8 (29.8 mg, 17.2 μmol) was dissolved in TEA/CH$_2$Cl$_2$ (0.15/2 mL), and TMSBr (44 μL, 346 μmol) was added into the solution. The reaction mixture was stirred at room temperature for 12 h. After this reaction completed, the solvent was removed by vacuum distillation and 5 mL of methanol was added to the resulting crude. After the dissolved crude was stirred at room temperature for another 12 h, the solvent was evaporated by vacuum. The residue was dissolved in CHCl$_3$ and washed by dilute aqueous HCl solution. The organic layer was separated and dried over MgSO$_4$, then the solvent was removed in vacuo. Furthermore, the residue was dissolved in MeOH and washed by hexane. The MeOH layer was separated and concentrated to yield 4’ as a red solid (26.5 mg, 15.8 μmol, 92%). Surface immobilizations were carried out without further purifications. HR–ESI MS: (m/z) 1673.6056 ([M–H]$^-$, C$_{79}$H$_{110}$FeO$_3$P, calcd. 1673.6016).

$^1$H NMR (500 MHz, CDCl$_3$, r.t.), δ: 7.86 (dd, $J = 11.6$, 8.1 Hz, 2H, ArH), 7.56 (d, $J = 5.9$ Hz, 2H, ArH), 7.43 (d, $J = 7.6$ Hz, 1H, ArH), 7.07–7.04 (m, 2H, ArH), 5.22–3.03 (m, 102H, CD–H, OCH$_3$, FcH). $^{13}$C NMR (126 MHz, CDCl$_3$, r.t.), δ: 158.80, 133.22, 130.80 (d, $J = 10.1$ Hz), 130.71 (d, $J = 13.9$ Hz), 124.61, 124.09, 123.93, 114.58, 114.78, 100.62, 100.25, 100.18, 100.16, 100.13, 99.73, 95.40, 90.06, 86.54, 85.92, 82.87, 82.51 (peaks overlapped), 82.46, 82.21, 82.16 (peaks overlapped), 81.91, 81.70, 81.26, 81.20 (peaks overlapped), 81.15 (peaks overlapped), 80.96, 71.94, 71.78, 71.52 (peaks overlapped), 71.47 (peaks overlapped), 71.33 (peaks overlapped), 71.29 (peaks overlapped), 71.21, 70.58, 69.94, 69.00, 67.63, 64.92,
61.88 (peaks overlapped), 61.83 (peaks overlapped), 61.77, 61.71, 59.33, 59.20, 59.11 (peaks overlapped),
59.09, 58.35, 57.91, 57.81 (peaks overlapped), 57.78, 57.33. $^{31}$P NMR (202 MHz, CDCl$_3$, r.t.), δ: 12.55.

Synthesis of 4

Under a nitrogen, S6 (34.6 mg, 20 μmol) was dissolved in TEA/CH$_2$Cl$_2$ (0.17/2 mL), and TMSBr (77 μL,
606 μmol) was added into the solution. The reaction mixture was stirred at room temperature for 12 h. After
this reaction completed, the solvent was removed by vacuum distillation and 5 mL of methanol was added
to the resulting crude. After the dissolved crude was stirred at room temperature for another 12 h, the solvent
was evaporated by vacuum. The residue was dissolved in CHCl$_3$ and washed by dilute aqueous HCl solution.
The organic layer was separated and dried over MgSO$_4$, then the solvent was removed in vacuo. Furthermore, the residue was dissolved in MeOH and washed by hexane. The MeOH layer was separated
and concentrated to yield 4 as a red solid (28.5 mg, 17 μmol, 85%). Surface immobilizations were carried
out without further purifications. HR–ESI MS: (m/z) 1673.6048 ([M–H]$^-$, C$_{79}$H$_{110}$FeO$_{33}$P, calcd. 1673.6016).

$^1$H NMR (500 MHz, CDCl$_3$, r.t.), δ: 8.15 (broad, 2H, ArH), 7.99 (dd, $J = 13.2$, 8.1 Hz, 2H, ArH), 7.43 (d, $J$
= 7.8 Hz, 1H, ArH), 7.25 (d, $J = 8.5$ Hz, 2H, ArH), 5.07–2.85 (m, 102H, CD–H, OCH$_3$, Fc–H). $^{13}$C NMR
(126 MHz, CDCl$_3$, r.t.), δ: 161.83, 132.98, 131.90 (d, $J = 15.1$ Hz), 131.05 (d, $J = 10.1$ Hz), 126.24, 126.14,
124.71, 115.76, 100.67, 100.13, 100.06, 99.98, 99.90, 97.93, 94.41, 91.90, 88.88, 84.72, 83.82, 82.70, 82.55,
82.44, 82.29, 82.10, 82.04, 81.89, 81.75, 81.68, 81.54, 81.28, 81.15, 81.09, 80.86, 80.64, 80.49, 72.47,
72.06, 71.80, 71.55 (peaks overlapped), 71.48, 71.45, 71.33, 71.17, 71.12, 70.60, 70.56, 70.21, 70.02, 69.28,
64.25, 61.86, 61.82, 61.74, 61.60, 61.30, 61.26, 59.13, 59.09, 58.94, 58.80, 58.76, 58.19, 57.89, 57.82,
57.73 (peaks overlapped), 57.69. $^{31}$P NMR (202 MHz, CDCl$_3$, r.t.), δ: 17.47.

Synthesis of 5'

Under a nitrogen, S13 (27.7 mg, 17.0 μmol) was dissolved in TEA/CH$_2$Cl$_2$ (0.23/2.5 mL), and TMSBr (71
μL, 554 μmol) was added into the solution. The reaction mixture was stirred at room temperature for 12 h.
After this reaction completed, the solvent was removed by vacuum distillation and 5 mL of methanol was
added to the resulting crude. After the dissolved crude was stirred at room temperature for another 12 h, the solvent was evaporated by vacuum. The residue was dissolved in CHCl₃ and washed by dilute aqueous HCl solution. The organic layer was separated and dried over MgSO₄, then the solvent was removed in vacuo. Furthermore, the residue was dissolved in MeOH and washed by hexane. The MeOH layer was separated and concentrated to yield 5' as a yellow solid (26.3 mg, 16.8 μmol, 99%). Surface immobilizations were carried out without further purifications. HR–ESI MS: (m/z) 1566.6326 ([M–H], C₇₄H₁₀₅NO₃₃P, calcd. 1566.6306). ¹H NMR (500 MHz, MeOD, r.t.), δ: 8.64 (broad, 2H, PyH), 7.84–7.80 (m, 2H, ArH), 7.75–7.73 (m, 2H, ArH), 7.67 (d, J = 4.0 Hz, 2H, PyH), 7.52 (d, J = 7.8 Hz, 1H, ArH), 7.27 (s, 1H, ArH), 7.23 (d, J = 7.9 Hz, 1H, ArH), 5.13–2.97 (m, 93H, CD−H, OCH₃). ¹³C NMR (126 MHz, MeOD, r.t.), δ: 172.96, 160.70, 148.63, 135.59, 134.26, 132.53 (d, J = 13.9), 132.09 (d, J = 10.1 Hz), 127.83, 126.62 (d, J = 191.5 Hz), 124.09, 116.81, 115.61, 101.38, 100.93, 100.89, 100.45, 100.33, 100.28, 100.22, 97.21, 96.17, 88.45, 83.73, 83.46, 83.40, 83.36, 83.33, 83.26, 83.08 (peaks overlapped), 83.02, 82.97, 82.87, 82.80 (peaks overlapped), 82.69, 73.35, 73.32, 73.23, 73.20, 72.98, 72.91, 72.60, 72.54, 72.45, 72.37, 71.74, 69.26, 62.31, 62.28, 62.15, 62.12, 62.04, 62.01, 59.57, 59.42, 59.35 (peaks overlapped), 59.32, 59.10, 58.66, 58.53, 58.44, 58.43, 58.08, 49.00. ³¹P NMR (202 MHz, MeOD, r.t.), δ: 14.60.

Preparation of Modified ITO Substrates

ITO substrates (surface roughness: average 0.7 nm (KURAMOTO)) or single crystalline ITO substrates (atomically flat terraces and steps (<0.2 nm))₆₀,₆¹ was used. Spectroscopic grade MeOH and CHCl₃ were used as solvents. Prior to use, the ITO surface was annealed with ozone at 200 °C for 15 min. An annealed ITO substrate was immersed in a methanol solution of the molecules for 12 h at room temperature, followed by rinsing in MeOH as well as chloroform and drying through nitrogen flow.

Electrochemical Measurement

Cyclic voltammetry (CV) measurements were taken using a CHI 600B potentiostat with a one-compartment cell under air. All measurements were carried out in 0.1 M of tetrabutylammonium hexafluorophosphate...
(TBAPF₆) in CH₂Cl₂. The 4–modified ITO substrate and 4’–modified ITO substrate were used as the working electrode (electrode area: 3.5² π mm²), and a platinum wire was used as the counter electrode. Potentials are referenced to an Ag/Ag⁺ (10 mM AgNO₃ in 0.1 M TBAClO₄–MeCN) electrode.

Electrochemical hydrogen peroxide (H₂O₂) production

Electrochemical H₂O₂ production was performed using a conventional three-electrode cell using Co²⁺(Ch)−5/FTO or Co²⁺(Ch)−5'/FTO, an Au coil counter electrode and a Ag/AgCl reference electrode in an oxygen (O₂)–saturated aqueous solution (8 mL) in an aqueous perchloric acid (pH 1.3, 0.1 M, 11.0 mL) by applying constant voltage (E = 0.41 V vs. Ag/AgCl) to a working electrode. The cell was kept in dark to prevent the unexpected decomposition of produced H₂O₂. The O₂ bubbling was continued during the photocatalytic reaction. The concentration of H₂O₂ produced in the reaction solution was determined by spectroscopic titration with an acidic solution of oxo[5,10,15,20-tetra(4-pyridyl)porphyrinato]titanium(IV) complex (Ti(TPyP)). The stock solution of Ti(TPyP) was prepared by dissolving 3.40 mg of Ti(TPyP) in 100 mL of 50 mM hydrochloric acid. An aliquot of the reaction solution was sampled and diluted with a certain amount of water depending on the concentration of H₂O₂. The diluted H₂O₂ solution was mixed to 0.25 mL of 4.8 M an aqueous solution of HClO₄ (4.8 M, 0.25 mL) and the stock solution of Ti(TPyP) (0.25 mL). The mixed solution was then allowed to stand for 5 min at room temperature. The sample solution was diluted to 2.5 mL with water and used for the UV–Vis absorption analysis to determine the absorbance at λ = 450 nm. A blank solution was prepared in a similar manner by adding distilled water instead of the sample solution to Ti(TPyP) in the same volume with its absorbance designated as A_B. The difference in absorbance was determined as follows: ΔA₄₅₀ = A_S – A_B. The amount of H₂O₂ produced was determined based on ΔA₄₅₀ and the volume of the solution.

Data Availability

All other data that support the findings of this study are available within the article and its Supplementary Information, or from the corresponding author upon reasonable request.
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**Ethics declarations**

**Competing interests**

The authors declare no competing interests.

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