Rare earth monopnictides and monochalcogenides from first principles: towards an electronic phase diagram of strongly correlated materials

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\textbf{Abstract.} We present the results of an \textit{ab initio} study of the electronic structure of 140 rare earth compounds. Specifically, we predict an electronic phase diagram of the entire range of rare earth monopnictides and monochalcogenides, composed of metallic, semiconducting and heavy fermion-like regions and exhibiting valency transitions brought about by a complex interplay between ligand chemistry and lanthanide contraction. The calculations exploit the combined effect of a first-principles methodology, which can adequately describe the dual character of electrons, itinerant versus localized, and high-throughput computing made possible by the increasing available computational power. Our findings, including the predicted ‘intermediate valent’ compounds SmO and TmSe, are in excellent overall agreement with the available experimental data. The accuracy of the approach, proven e.g. through the lattice parameters calculated to within \(\sim 1.5\%\) of the experimental values, and its ability to describe localization phenomena in solids, makes it a competitive atomistic simulation approach in the search for and design of new materials with specific physical properties and possible technological applications.
1. Introduction

Increased computer power, together with reliable electronic structure codes, has opened up the possibility of exploring the electronic properties of a large number of materials [1]. A systematic first-principles-based theoretical study of entire families of compounds has the potential for exploring new trends in their physical properties or discovering materials with improved characteristics that are of possible interest for technological applications.

The ab initio determination of material properties requires electronic structure calculations that are accurate and parameter free. With the advent of density functional theory (DFT) [2], quantum mechanical calculations of the electronic structure of solids were put on a rigorous theoretical foundation. Nevertheless, the lack of knowledge of the exact exchange-correlation energy functional constitutes a substantial obstacle to the predictive power of the calculations, as different mechanisms are at work depending on the relative importance of band formation energy on the one hand and on-site electron–electron interaction and electron localization on the other. Standard electronic structure calculations, based on the local spin density approximation (LSDA) [3, 4] to DFT, work very well for conventional materials such as simple metals and their alloys, characterized by delocalized electron states and large band formation energies. However, for systems where localized electron states occur, such as the 4f’s in rare earth compounds, the LSDA suffers from a sizeable self-interaction error [5] brought about by the local approximation to the exchange-correlation energy functional. Consequently, different assumptions and/or parameters need to be invoked to describe the electronic structure of strongly correlated electron materials [6].

The self-interaction-corrected local spin density (SIC-LSD) [7] energy functional is free of the self-interaction error and fulfils the requirement of being parameter free and of sufficient accuracy to study trends in the properties of entire families of materials containing both localized and itinerant electrons [8]–[10]. Here we apply the SIC-LSD to provide global understanding of the ground state electronic structure of rare earth monopnictides and monochalcogenides and, through that, also demonstrate the predictive power and accuracy
of this approach. Specifically, the present study covers 140 distinct compounds, RX, with R referring to the rare earth atoms Ce, Pr, Nd, Pm, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, Yb, Lu and X standing for the pnictide atoms N, P, As, Sb, Bi and the chalcogenide atoms O, S, Se, Te, Po. Most of these compounds are known to exist in nature, crystallizing predominantly in the rocksalt structure. Due to the intricate dual nature of the 4f-electrons, as a function of chemical environment these compounds display an extraordinarily wide range of electronic, magnetic, optical and magneto-optical properties with, among other things, potential practical applications in the field of spintronics [11, 12]. There has been a considerable number of theoretical studies on the RX compounds and this has resulted in greatly improved insights into their properties [13]–[16]. However, those calculations have often been guided by experiment, relying on parameters to reproduce the observed properties. The goal of the present systematic study is to establish a parameter-free unified theoretical picture of the RX electronic properties and possibly provide directions for further experimental investigations. With future applications in mind, we show how large-scale computational infrastructures can potentially be exploited in the search for novel rare earth-related materials that play an increasingly prominent role in innovative technologies and the quest for renewable energies [17].

2. Theoretical background

In the LSD approximation the total energy functional is decomposed as

$$ E^{\text{LSD}} = \sum_\alpha \langle \psi_\alpha | \hat{T} | \psi_\alpha \rangle + U[n] + V_{\text{ext}}[n] + E^{\text{LSD}}_{\text{xc}}[n, n], $$

where $\hat{T}$ is the kinetic energy operator and $V_{\text{ext}}[n]$ represents the interaction with an external potential. The electron–electron interaction is given as the sum of the Hartree term $U[n]$ and the local exchange-correlation term, $E^{\text{LSD}}_{\text{xc}}[n, n]$. In the Hartree–Fock approximation, the expectation value of the electron repulsion operator in some $f^n$ multiplet state \( \Psi^n(\mathbf{L}, \mathbf{S}, \mathbf{J}) \) may be written as [18, 19]

$$ \langle \Psi^n(\mathbf{L}, \mathbf{S}, \mathbf{J}) | \sum_{\mathbf{r}_{ij}} \frac{1}{\mathbf{r}_{ij}} | \Psi^n(\mathbf{L}, \mathbf{S}, \mathbf{J}) \rangle = \frac{1}{2} n(n - 1) E^0 + \alpha_1 E^1 + \alpha_2 E^2 + \alpha_3 E^3, $$

where $\Psi^n(\mathbf{L}, \mathbf{S}, \mathbf{J})$ is, in general, given by a linear combination of $n \times n$ Slater determinants. Since $\sum (1/\mathbf{r}_{ij})$ commutes with total $\mathbf{L}$, $\mathbf{S}$ and $\mathbf{J}$, the above matrix elements are equal for $M_J$ quantum numbers belonging to one ($\mathbf{L}, \mathbf{S}, \mathbf{J}$) family. $E^i$ are the Racah parameters [20], which are related to the Slater two-electron integrals (defined later). The coefficients $\alpha_i$ depend on the particular $\mathbf{L}, \mathbf{S}, \mathbf{J}$-configuration chosen.

The quantity $E^0$ is the Coulomb repulsion between two $f$-electrons, and the Coulomb energy term in the LSD total energy functional (1) corresponds to the $\frac{1}{2} E^0 n^2$ part of the first term in equation (2). The term $-(1/2) E^0 n$ in equation (2) cancels an equivalent contribution in the Hartree energy term, representing the interaction of an electron with itself. A similar cancellation does not occur in the LSD functional, which therefore carries an unphysical self-interaction [7]. This interaction, which tends to be insignificant for extended band states, may lead to uncontrollable errors in the description of atomic-like localized states, for example the $f$-electrons in the rare earths. Following the suggestion of Perdew and Zunger [7], in the SIC-LSD methodology, the LSD functional (1) is corrected for this spurious self-interaction.
by adding an explicit energy contribution for an electron to localize. The resulting orbital-dependent SIC-LSD total energy functional has the form \( E_{\text{SIC-LSD}} = E_{\text{LSD}} + E_{so} - \Delta E_{\text{sic}} \), where

\[
\Delta E_{\text{sic}} = \sum_{\alpha} \delta_{\alpha}^{\text{sic}} = \sum_{\alpha} \left\{ U[n_{\alpha}] + E_{\text{xc}}^{\text{LSD}}[\bar{n}_{\alpha}] \right\},
\]

\[
E_{so} = \sum_{\alpha} \langle \psi_{\alpha} | \xi(\vec{r}) \hat{l} \cdot \vec{s} | \psi_{\alpha} \rangle.
\]

The self-interaction energy (4) consists of the self-Coulomb and self-exchange-correlation energies of the occupied orbitals \( \psi_{\alpha} \) with charge density \( n_{\alpha} \) and spin density \( \bar{n}_{\alpha} = (n_{\alpha}^{\uparrow}, n_{\alpha}^{\downarrow}) \). For itinerant states, the self-interaction \( \delta_{\alpha}^{\text{sic}} \) vanishes identically, whereas for localized (atomic-like) states, \( \delta_{\alpha}^{\text{sic}} \) may be appreciable. Thus, the self-interaction correction constitutes a negative energy contribution gained by an electron upon localization, which competes with the band formation energy gained by the electron if allowed to delocalize and hybridize with the available conduction states. Different localized/delocalized configurations are realized by assuming different numbers of localized states—here \( f \)-states on rare earth atom sites. The SIC-LSD approach is fully \textit{ab initio}, as both localized and delocalized states are expanded in the same set of basis functions and are thus treated on an equal footing. If no localized states are assumed, \( E_{\text{SIC-LSD}} \) coincides with the conventional LSD functional; that is the Kohn–Sham minimum of the \( E_{\text{LSD}} \) functional is also a local minimum of \( E_{\text{SIC-LSD}} \).

The spin–orbit interaction (5) couples the band Hamiltonian for the spin-up and spin-down channels; that is, a double secular problem must be solved. The spin–orbit parameter

\[
\xi(r) = -\frac{2}{c^2} \frac{dV}{dr},
\]

in atomic Rydberg units, is calculated from the self-consistent potential \( V \).

Of the remaining terms in equation (2), the \( E^{1} \) term accounts for Hund’s first rule. The \( E^{2} \) term only contributes to the level spacing of excited multiplets and will not be relevant to a functional for the ground state energy. The \( E^{3} \) term describes the level spacing between multiplets in the maximum spin configuration; that is, this term accounts for Hund’s second rule, by which the multiplet of maximum total orbital momentum, \( L \), has the lowest energy. With respect to the average \( f^{n} \) (maximum spin) energy (the grand barycentre), the maximum \( L \) multiplet is lowered by \(-jE^{3}\), where \( j = (0, 9, 21, 21, 9, 0, 0) \) for \( n = (1–7) \), and \( n = (8–14) \), respectively. It is an atomic effect that results in increased stability at 1/4 and 3/4 filling of the 4\( f \)-shell, and this second Hund’s rule effect is often referred to as the tetrad effect (TE) \cite{19}. Although both LSD and SIC-LSD take into account Hund’s first and third rules, with respectively the exchange interaction and spin–orbit coupling included in the total energy functional, Hund’s second rule is not defined for the homogeneous electron gas, underlying LSDA \cite{21}. To account for it in our total energy calculations, we add \textit{(a posteriori)} to the SIC-LSD total energy functional the relevant correction

\[ \Delta E_{i} = -jE^{3}. \]
The $E^3$ parameter is equivalent to Racah’s $B$-parameter \[20\] and is given in terms of the reduced Slater integrals $F_i$ as

$$E^3 = \frac{1}{3}(5F_2 + 6F_4 - 91F_6),$$

where $F_k = F^k / D^k$. Here $D^k$ are numerical constants ($D^k = 225, 1089, 7361.64$ for $k = 2, 4, 6$) and the Slater integral $F^k$ is defined by

$$F^k = e^2 \int_0^\infty \int_0^\infty \frac{(r_\leq)^k}{(r_\geq)^{k+1}} \phi_{4f}^2(r_1)\phi_{4f}^2(r_2) r_1^2 r_2^2 dr_1 dr_2.$$

Here, $\phi_{4f}$ is the $f$-partial wave as calculated in the self-consistent crystal potential, and $r_\leq$ ($r_\geq$) denotes the smaller (larger) of the variables $r_1$ and $r_2$. In reality, correlation effects in the solid environment tend to reduce the multiplet energy level splitting. Hence, our calculated TE values are on average 15–20% larger than their experimental counterparts \[19\].

3. Calculational details

Given the total energy functional $E^{\text{SIC-LSD}}$, the computational procedure is the same as for the LSD case, i.e. minimization is accomplished by iteration until self-consistency. In this work, the electron wavefunctions are expanded in the linear-muffin-tin-orbital (LMTO) basis functions \[22\]. The atomic spheres approximation (ASA) is used whereby the crystal volume is divided into slightly overlapping atom-centred spheres of total volume equal to the actual volume. A known shortcoming of the ASA is that different crystal structures have different degrees of overlap of the ASA spheres, resulting in substantial relative errors in the evaluation of the total energy. Although this inhibits the comparison of the energies of different crystal structures, when comparing the energies of different localization/delocalization scenarios within the same crystal structure the ASA error is of minor influence. To improve the packing of the structure, empty spheres have been introduced on high-symmetry interstitial sites. Two uncoupled energy panels have been considered when constructing the LMTOs to ensure an accurate description of semicore states. The $LS$-coupling scheme is adopted for the localized $f$-states by starting the iterations with Wannier states of appropriate symmetry. During iteration to self-consistency the symmetry of the Wannier states may change, but they grossly retain their overall characteristics, because the energy scale of spin–orbit interaction is smaller than that of exchange, but larger than that of the crystal field for the $f$-states. The rocksalt structure with the magnetic moments arranged ferromagnetically was assumed in all the calculations. To conduct the present large-scale systematic study of rare earth compounds, in addition to a reliable theory and computer codes we have made extensive use of grid computing and automated data/metadata management tools \[23, 24\].

4. Total energies and valency transitions

In the SIC-LSD methodology, both the itinerant and localized limits are described by the same energy functional, which enables us to predict the ground state of correlated electron materials from total energy considerations \[8, 9\]. Hence, one can investigate localization phenomena in solids \[6\] by realizing and studying different localization/delocalization scenarios, giving rise to different valency configurations, and through total energy minimization determine the ground state electronic structure and valency configurations of the compound under consideration \[25\].
This nominal valency is defined as an integer number of rare-earth electrons available for band formation,
\[ N_{\text{val}} = Z - N_{\text{core}} - N_{\text{SIC}}, \] (9)

obtained by subtracting from the atomic number \( Z \) the sum of core \( N_{\text{core}} \) and localized \( N_{\text{SIC}} \) electrons. In this paper, we will be using two interchangeable nomenclatures, \( f^n \) and \( R^m^+ \), to describe the configuration of the rare earth ion, implying \( n = N_{\text{SIC}} \) and \( m = N_{\text{val}} \), respectively. The total number of \( f \)-electrons may be larger than \( n \), because, in addition to the \( n \) localized \( f \)-states, the band states also contribute to the total \( f \)-count on a given ion. Note that our calculated valencies refer to the number of rare earth electrons that contribute to bonding and thus do not necessarily coincide with the nominal (ionic) valency of a compound.

In our study of the rare earth monopnictides and monochalcogenides, for each given compound the self-consistent calculations involve the total energy minimization as a function of lattice parameter with respect to a number of different possible localization/delocalization scenarios, in order to determine the global energy minimum and the corresponding equilibrium volume. To understand what is involved in this process, we explain the minimization procedure on the example of HoSe depicted in figure 1(a). What is seen in the figure are total energy curves, as a function of lattice parameter, for three energetically relevant valency configurations. The way they come about is based on electronic structure considerations. The Ho atom contributes 67 electrons to the total electron count of HoSe, 54 of which are considered to be part of the core (and semicore) and thus not contributing significantly to its chemistry. The remaining 13 electrons on the outermost shells, 11 of which are the \( f \)-electrons, are chemically active and...
constitute the valence electrons. The latter can either hybridize with the other band electrons or localize on site, playing no role in band formation. Assuming all 11 $f$-electrons to be localized means that only two $s$-electrons of Ho contribute to band formation, a scenario that we refer to as divalent and that represents, with respect to $f$-electrons, the fully localized limit. Configurations with one or two delocalized (bonding) $f$-electrons are, respectively, referred to as trivalent and tetravalent. The LDA scenario, which treats all the ligand and rare earth valence electrons, including the 11 $f$-electrons, as band electrons, is found to be energetically unfavourable by $\sim 1$ Ryd, as compared to any of the other scenarios, and is therefore not shown in figure 1(a). Altogether, at least some 40 self-consistent calculations are required for establishing the global energy minimum for a compound like HoSe that, as can be seen in figure 1(a), is obtained in the trivalent scenario.

From the calculated local energy minima, we can evaluate the energy differences between respectively the divalent and trivalent configurations, $\Delta E_{\text{II} - \text{III}}$, and the tetravalent and trivalent configurations, $\Delta E_{\text{IV} - \text{III}}$, both of which are a measure of the stability of the ground state configuration. From figure 1(a), we find for HoSe $\Delta E_{\text{IV} - \text{III}} = 90$ mRyd and $\Delta E_{\text{II} - \text{III}} = 20$ mRyd, indicating a rather stable trivalent ground state, with no tendency to delocalization of an additional $f$-electron, but with some, albeit marginal, tendency to localization of an additional electron. Thus, while we predict a trivalent ground state in ambient conditions, with respect to possible phase transformations, for example under negative pressure, $\Delta E_{\text{II} - \text{III}}$ is the relevant energy scale to consider in HoSe. The calculated trivalent equilibrium lattice parameter is equal to 5.55 Å, differing from the divalent minimum by about 5%.

In figure 1(b), we present the calculated total energy differences, $\Delta E_{\text{II} - \text{III}}$, for the entire rare earth selenide series (RSe) from Pr onwards. The two curves shown refer, respectively, to the SIC-LSD calculations with (stars) and without (squares) the TE. The positive values indicate a trivalent ground state, whereas the negative values imply that the divalent configuration is energetically more favourable. One can see that without the TE, $\Delta E_{\text{II} - \text{III}}$ decreases monotonically from trivalent PrSe to divalent EuSe, and again from trivalent GdSe to divalent YbSe. The increasing trend towards divality is associated with the enhanced localized nature of the $f$-electrons and can be understood as a manifestation of the well-known lanthanide contraction. A small $\Delta E_{\text{II} - \text{III}}$ indicates that for the given compound no clear ground state emerges, as the divalent and trivalent configurations are close to energetically degenerate, due to the competition between band formation and self-interaction correction (localization) energies.

The overall trend of decreasing energy differences is still recognizable in both halves of the selenide series when the TE is included. However, due to the abrupt decrease in $\Delta E_{\text{II} - \text{III}}$ at DySe a characteristic hump, also observed in experiments on rare earths [26, 27], develops in the latter part of the series, implying that this compound is considerably less trivalent than one would expect from the SIC-LSD result prior to taking Hund’s second rule into account. Even more important are the changes in predicted ground state configurations for ErSe and TmSe (as well as PmSe), indicating that the TE can bring about a qualitative change in compounds where the original energy differences are small. Otherwise, implementing Hund’s second rule has only a quantitative effect on the $\Delta E_{\text{II} - \text{III}}$ of those compounds that from the outset are either solidly trivalent or divalent.

Taking into account the TE, the divalent–trivalent energy differences, $\Delta E_{\text{II} - \text{III}}$, for all the RX compounds (except for the rare earth nitrides and the Ce and Lu compounds) are collected in figure 2. The positive and negative values again indicate respectively trivalent and divalent ground state configurations. In figure 2(a), for each of the ligand ions the $\Delta E_{\text{II} - \text{III}}$ data are...
Figure 2. (a) The energy difference $\Delta E_{\text{II} - \text{III}}$ as a function of ligand ion for all the rare earths. (b) The energy difference $\Delta E_{\text{II} - \text{III}}$ as a function of rare earth ion for all the ligands. The energy differences are given in mRyd.

plotted as clusters of bars, representing the rare earth ions ranging from Pr to Yb. We note that the total energy trends, observed for the selenide series in figure 1(b), are reproduced in all the ligands, but overall in the chalcogenides the divalent configuration becomes increasingly more favourable as a function of ligand size, while in the pnictides the trivalent configuration remains the ground state, even for the ‘late’ rare earths Eu and Yb (with the exception of EuBi). The trivalent-to-divalent localization transition thus mainly pertains to the rare earth chalcogenides.

In figure 2(b), the same total energy data are depicted as a function of rare earth, with the ligands plotted as clusters of bars ranging from P to Po for each of the rare earth ions. In this representation, one can clearly see the combined effect of the lanthanide contraction, leading gradually to the more preferred divalent configurations, and the repetition of this trend due to the half-filled shell effect.
Figure 3. The valency phase diagram for the entire manifold of rare earth monopnictides and monochalcogenides. The blue areas mark divalent compounds, while trivalency is indicated by the yellowish colour. The red parts of the diagram refer to the close-to-tetravalent compounds (mostly Ce compounds). The light blue/white regions represent the borderline cases between divalent and trivalent configurations. The colour scheme refers to the valencies, obtained by scaling the calculated energy differences such that a value of 3.5 indicates trivalent–tetravalent degeneracy as observed in CeN, while values smaller than 2.5 refer to divalent ground state configurations.

Although $\Delta E_{\text{II} - \text{III}}$ is the relevant energy scale for the large majority of the RX, there are a number of exceptions where the delocalized tetravalent scenario is energetically more favourable than the localized divalent scenario even though according to our calculations it never actually becomes the true ground state. The trivalent–tetravalent energy difference, $\Delta E_{\text{IV} - \text{III}}$, is the more relevant energy scale for the Ce compounds for example, as the divalent configuration is always very unfavourable, and in most cases could not even be converged. In the case of CeN, and to some degree in CeO, the tetravalent and trivalent scenarios are actually close to energetically degenerate. Since the divalent scenario is so highly unfavourable, the Ce pnictides and chalcogenides, as well as all the other rare earth nitrides, are not shown in figure 2. For the LuX compounds $\Delta E_{\text{II} - \text{III}}$ is not adequate, as a divalent scenario would here imply 15 localized $f$-electrons.

As shown in figure 3, a unified picture for all the RX compounds emerges when we map the calculated energy differences into ground state valencies. Here the valency scale is defined such that it runs from 3.5 for the tetravalent/trivalent degenerate compounds ($E_{\text{IV} - E_{\text{III}} = 0}$) to 2.5 for the trivalent/divalent degenerate ($E_{\text{II} - E_{\text{III}} = 0}$) compounds. The yellow/orange areas, which dominate throughout the phase diagram, demonstrate the overall preferred trivalent ground state configurations.
configuration for a large number of RX compounds, in particular the monopnictides. The trend towards delocalizing an additional electron is only noticeable for the CeX systems and some of the nitrides, as indicated by the more reddish areas of the phase diagram, while the trivalent-to-divalent localization transition occurs around the regions indicated in white. The blue zones represent the stable divalent configuration, which is the dominating ground state for many of the ‘heavy’ chalcogenides, in combination with the rare earths situated in the middle and latter parts of the series.

5. Electronic structure

5.1. Density of states: chalcogen versus pnictogen chemistry

The observed total energy trends can be understood in terms of variations in the ground state electronic structures brought about by changes in the atomic composition, the distance between atoms and the relative position of the atomic energy levels. In the RX compounds, the electronic structure results mainly from the $f$- and $d$-orbitals on the R-sites overlapping with the $p$-orbitals on the ligand sites, as well as with the $f$- and $d$-orbitals on neighbouring R-sites. In figure 4, using TbP (column 1) and TmTe (column 2) as representative examples of the pnictide and chalcogenide rare earth compounds, we depict the density of states (DOS) for three different valency scenarios, as realized, respectively, in the delocalized (figures 4(a) and (a')), trivalent (figures 4(b) and (b')) and divalent (figures 4(c) and (c')) configurations.

The LSD band picture, for both TbP (figure 4(a) and TmTe (figure 4(a')), is characterized by a large, $f$-derived, DOS at the Fermi level. A noticeable difference is observed with respect to the relative strength of the exchange and spin–orbit interactions. In TbP the exchange interaction dominates, which results in the splitting of the $f$-states into majority spin states, situated at the bottom of the phosphorus $p$-band, and minority spin states, situated at the Fermi level. In TmTe the spin–orbit interaction has become more important, splitting the thulium $f$-states into $j = 5/2$ and $j = 7/2$ bands, with the exchange interaction being responsible for the additional splitting within each of the two groups of bands.

In spite of its interesting features, the band picture (figures 4(a) and (a')) is an unphysical representation of the RX compounds, as the large DOS at the Fermi level is not observed experimentally. The trivalent scenarios of figures 4(b) and (b') are obtained by localizing, respectively, eight $f$-electrons in TbP (Tb$^{3+} \equiv \text{Tb}(f^8)$) and 12 $f$-electrons in TmTe (Tm$^{3+} \equiv \text{Tm}(f^{12})$). Compared to the LSD scenario the trivalent configuration is found to be energetically more favourable by $\sim 700 \text{ mRyd}$ for TbP and $\sim 1200 \text{ mRyd}$ for TmTe. In TbP, the seven majority spin states, as well as one of the minority spin states, are no longer available for band formation and hybridization and have vanished from the band picture in figure 4(b). Similarly, in the DOS for TmTe (figure 4(b')), the localization of the six $j = 5/2$ $f$-electrons as well as six of the $j = 7/2$ electrons implies that only two $f$-states are available for band formation. Note that in the DOS plots only the band states are displayed, i.e. the itinerant valence states, including the delocalized $f$-states. The localized $f$‘s are not shown since the SIC-LSD approach, which after all is a one-electron ground state theory, does not give accurate removal energies of localized states due to electron–electron interaction effects [28] and the neglect of screening and relaxation contributions [29].

The trivalent DOSs, in respectively figures 4(b) and (b’), differ noticeably from each other, as in the former the Fermi level is situated in the gap between the filled phosphorus $p$-band...
Figure 4. Left column: density of states of TbP, respectively, for the (a) LSD, (b) trivalent and (c) divalent Tb ion configuration. The majority (minority) spin components are displayed as positive (negative) values. Right column: density of states of TmTe, respectively, for the (a’) LSD, (b’) trivalent and (c’) divalent configurations of the Tm ion. Density of states: total (dotted black), ligand p-projected (solid blue), rare earth d-projected (solid green) and rare earth f-projected (solid red). The Fermi level is indicated by the dashed vertical line.

and the unoccupied $f d$-band, while in the latter the Fermi energy is pinned to the sharp $f$-peak above the conduction band minimum. The P atom has three unoccupied $p$-orbitals and in the TbP crystal it can accommodate three electrons from the Tb ion through charge transfer and hybridization, which results in the filled $p$-band of figure 4(b). The Te atom, on the other hand, has one additional $p$-electron in its outer shell, which is why the $p$-band in TmTe can only accommodate two electrons, and as a consequence of this, here in the trivalent Tm scenario, the third electron is forced to occupy the $f d$-conduction band.

The variation in $p$-orbital occupation is fundamental to the overall difference we observe in figure 3 with respect to the ground state valency configurations of the pnictides and chalcogenides. Thus we find that when trying to localize an additional $f$-electron in TbP, the resulting divalent configuration, depicted in figure 4(c), is energetically unfavourable compared to the trivalent scenario in figure 4(b) by $\sim 250$ mRyd. The $f^8$ to $f^9$ localization is associated with charge transfer, and brings about the depopulation of the P $p$-states, as indicated by the Fermi level being situated in the $p$-band. The removal of the bonding states results in a considerable loss in hybridization and Madelung energy, which is not compensated for by the gain in self-interaction energy, and the divalent scenario will therefore not be realized in TbP. In

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Figure 5. Schematic representation of DOSs for the rare earth monopnictides (upper panel) and monochalcogenides (lower panel). Each DOS is composed of a number of rectangles representing respectively the ligand $p$-states (situated lowest in energy), the rare earth $d$-states (situated highest in energy) and the occupied and unoccupied $f$-states (upright bars). Occupied states are shown in blue. For each panel, the three different scenarios ((A) to (C) and (D) to (F)) represent the different trivalent DOSs resulting from the decrease in electronegativity and the increase in $f$-state localization as one moves along the diagonal of the corresponding compound space. Scenarios (C') and (F') refer to the divalent scenarios obtained respectively from (C) and (F) through localizing an additional $f$-electron.

TmTe on the other hand, the localization transition from $f^{12}$ in figure 4(b') to $f^{13}$ in figure 4(c') is associated with a minor loss in band formation energy, given the vanishing width of the occupied $f$-peak, and the gain in self-interaction (localization) energy is comparatively larger by $\sim 35$ mRyd. Accordingly, the divalent scenario of figure 4(c') becomes the ground state scenario for TmTe.

5.2. Electronegativity and lanthanide contraction

Apart from the fundamental difference between the pnictide and chalcogenide electronic structures, the observed energy trends in each series, as a function of R ion in figure 2(a) and as a function of ligand in figure 2(b), are determined by further factors such as ligand electronegativity and lanthanide contraction. These changes in electronic structure are presented schematically in figure 5. Here we depict the DOSs for eight distinct scenarios observed
throughout the rare earth compounds. Each DOS is composed of several rectangles, referring respectively to the ligand $p$-states (situated lowest in energy), the rare earth $d$-states and the localized and delocalized $f$-states. The occupied states are indicated in blue. For both pnictides and chalcogenides, the electronegativity decreases with increasing anion size, i.e. respectively from N to Bi and from O to Po. On the other hand, the $f$-electron localization increases due to the lanthanide contraction, as we move through the rare earth series from Ce to Eu, and again from Gd to Yb. In terms of DOS, these two trends manifest themselves respectively via ligand $p$-bands moving up in energy (towards lower binding energies) and unoccupied $f$-states moving towards higher binding energies. In figure 5, the combined effect of these two trends is depicted in the two sets of DOSs respectively along the diagonals of the pnictide ((A) $\rightarrow$ (C)) and chalcogenide compounds ((D) $\rightarrow$ (F)). As the $f$-states move towards lower energy they get pinned to the Fermi level and start filling up. Ultimately the degree of $f$-band filling becomes such that the gain in energy associated with localizing the given $f$-state becomes more important than the corresponding loss in band formation energy, and the divalent scenario becomes the ground state (scenarios (C$'$) and (F$'$)) [30]. As seen in figure 5, owing to the additional electron, in the chalcogenides the filling of $f$-states sets in much earlier than in the pnictides, which explains why the trivalent scenario is relatively more dominant in the pnictides.

6. Electronic properties

6.1. Lattice parameters

For each given compound, the calculated global energy minimum determines the equilibrium lattice parameter as well as the ground state. The calculated lattice parameters for the rare earth sulfides and the ytterbium pnictides and chalcogenides are compared to their experimental counterparts in figures 6(a) and (b), respectively. The agreement is good, reproducing the sudden increase in lattice parameter at SmS, EuS and YbS, due to the increased $f$-electron localization. We have observed that quite systematically the calculated lattice parameters for the divalent and trivalent scenarios of a given compound differ by about 5%. Therefore, a comparison to the experimental lattice parameters is expected to provide a good indication of whether the calculations predict the correct ground state and valency configuration. In figure 7, the experimental equilibrium lattice parameters for all the rare earth monopnictides and monochalcogenides [31] are plotted along the $x$-axis, while our calculated equilibrium counterparts are plotted along the $y$-axis. In this representation, the proximity of the data points to the $x = y$ line is a measure of the agreement between theory and experiment. We find that for the large majority of the compounds the calculated equilibrium lattice parameters lie within $\sim 1.5\%$ of the experimental values, i.e. well below the 5% that would indicate a wrongly predicted ground state configuration. The only exceptions are CeN, DyTe and three of the polonide compounds. With respect to the latter compounds, polonium is difficult to handle, and it is unclear to what degree the experimental results refer to stoichiometric samples [32].

It is well established from rare earth halide data [33] as well as XPS measurements on the rare earth elements [26] that the divalent configuration becomes more competitive in Dy compounds due to the 3/4 filling of the $f$-shell. The inclusion of the TE leads to a considerable reduction of $E_{II} - E_{III}$ for DyX compounds, as can be seen on the example of DySe in figure 1(b), and quite generally gives rise to the negative dip around Dy in figure 2(b). This trend reproduces overall very well the experimentally observed data [27]; however, we wrongly predict a Dy$^{2+}$

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Figure 6. Lattice parameters of (a) the lanthanide sulfides (RS) and (b) the Yb pnictides and chalcogenides (YbX) as calculated in their ground state configuration.

ground state in DyTe. This seems to indicate that in some cases we tend to overestimate the TE’s influence as a result of approximating the respective Slater integrals by the fully localized atomic limit instead of considering also corrections due to the solid-state environment (screening/hybridization). For CeN, the experimental lattice parameter seems to suggest a tetravalent ground state [34], whereas our calculations predict the trivalent configuration to be marginally more favourable than the tetravalent configuration by $\sim 3$ mRyd, which on the other hand could be indicative of an intermediate valency scenario implied by experiments [35].

In figure 7, the wrongly predicted ground state of DyTe shows up as a noticeably off-diagonal data point, but at the same time it also highlights the excellent agreement obtained for
all the other cases investigated with the present approach. Thus, the SIC-LSD methodology is capable of predicting, with considerable accuracy, the ground state valency and lattice parameter of the rare earth monochalcogenides and monopnictides, and the associated ground state electronic structures can be expected to make valid predictions concerning the other physical properties of these materials.

6.2. Electronic phase diagram

Here we concentrate on establishing global trends in the ground state electronic structure, which we base on the calculated DOS at the Fermi energy, $E_F$, as providing a guideline for predicting materials with specific properties when we systematically scan the full manifold of rare earth compounds. The corresponding results are summarized in the form of an ‘electronic phase diagram’ presented in figure 8, where based on the calculated DOS we distinguish between semiconductors, semi-metals, metals and heavy fermion-like compounds.

Looking at the phase diagram, we see that our calculations predict a considerable number of compounds to be semiconductors. In particular, with respect to the pnictides, NdN, PmN and the majority of the rare earth nitrides from GdN onwards are found to be small-gap semiconductors, with energy band gaps ranging from 0.06 eV in NdN to 0.6 eV in HoN. Few conclusive conductivity measurements for the nitrides exist, and it is for example still not fully established whether GdN is semiconducting [36] or semi-metallic [37]. Our calculations predict GdN to be a semiconductor, although in an earlier SIC-LSD calculation, where spin–orbit coupling was not taken into account, a half-metallic ground state was observed instead [38]. All the divalent
Figure 8. A global picture of the electronic properties of all the studied pnictides and chalcogenides, based on the densities of states at the Fermi energy $E_F$. The colour scheme is as follows: metals are marked in blue tones, with light blue for semi-metals. Semiconductors are in black, while yellow to red is assigned to increasingly heavy fermion-like systems. The fields marked with ‘IV’ indicate intermediate behaviour between two different valency configurations.

Monochalcogenides turn out to be semiconductors, in good agreement with experiment. The divalent character implies that in the chalcogenide semiconductors the energy gap is brought about by localizing an additional $f$-electron, as depicted through the transition from (F) to (F') in figure 5, in variance to the energy gap observed in the trivalent nitride semiconductors, resulting from the filling of the unoccupied anion $p$-states, as represented by scenario (A) in figure 5. In the SIC-LSD the occupied 4$f$-levels tend to be situated at too low energies [39], as this ground state methodology cannot reproduce the spectroscopy of an eventual experimentally observed mid-gap $f$-level. For EuO, EuS and YbS, the calculations overestimate the energy gaps by a factor of 2 [40, 41].

The dominant part of the electronic phase diagram in figure 8 consists of metallic compounds. The indicated differences in conductivities can be straightforwardly related to the trends in electronic structure observed when moving along the diagonals in figure 5. In the pnictides, the closing of the semiconducting gap due to the onset of pnictogen $p$-rare earth $sd$ overlap gives at first rise to a vanishingly small DOS, resulting in the semi-metals CeX to PrX and GdX to DyX (light blue areas). As the overlap gradually increases (due to decreasing electronegativity and/or increasing lanthanide contraction), the behaviour becomes fully metallic (dark blue areas), and eventually in the ‘late’ rare earths, with more and more of the $f$-states getting occupied, heavy fermion-like behaviour sets in (yellow-red areas). The rare earth chalcogenides display less variety in terms of metallic behaviour. Here we start out with a partially occupied conduction band, and semi-metallic behaviour is not observed. Instead, as can be seen from figure 8, the rare earth systems CeX to PmX and GdX to ErX (apart from the polonides and a couple of tellurides) are metallic.
Compared to the pnictides, in the late rare earth chalcogenides, the $f$-states start filling even before they have moved far enough towards lower binding energies to achieve significant hybridization. Instead of forming a heavy metal, a relatively pure $f$-state pins the Fermi level (scenarios (E) and (F) in figure 5) and the divalent semiconducting scenario (scenario (F$'$) in figure 5) correspondingly becomes energetically more favourable.

The agreement between observed and predicted properties is overall very good, except maybe for the calculated heavy fermion pnictides. Thus the ‘late’ rare earths, SmX and YbX, are described as low carrier semi-metallic compounds, seemingly at odds with the large DOS at the Fermi level that is predicted in the SIC-LSD calculations. The experimentally observed large specific heat coefficient in these systems has been interpreted as heavy fermion behaviour by some authors [42], while others have explained it in terms of the Kondo effect [43, 44]. Since our calculations are based on the effective one-electron picture, they cannot adequately describe the many-body physics of these compounds. However, the large peak at the Fermi level that is calculated in the trivalent configuration indicates strong $pf$-mixing, i.e. a prerequisite for the hybridization between the narrow $f$-peak and the sea of conduction electrons that underpins the Kondo effect. EuSb, YbBi and EuBi (as well as EuAs) do not actually crystallize in the NaCl structure investigated here [45], i.e. the very same pnictides that we find to be divalent or close to divalent do not occur in nature. As indicated schematically in figure 5, for the pnictides, localization of an additional $f$-electron ((C) $\rightarrow$ (C$'$) transition) implies the removal of some bonding $p$-states. This, however, would destabilize the underlying NaCl crystal structure and, as a result of this competition between band formation and correlations, it seems that these materials prefer to distort and crystallize in a different structure altogether.

In figure 8, the compounds marked as ‘intermediate valent’ (IV) are those where the calculations find close to degenerate energy configurations, meaning that either $\Delta E_{\text{II} \rightarrow \text{III}}$ or $\Delta E_{\text{II} \rightarrow \text{III}}$ is smaller than 5 mRyd. Given the dual nature of the $f$-electrons, it is to be expected that these materials are highly sensitive to environmental changes such as pressure [46, 47] or doping. In the phase diagram of figure 3 they correspond to the trivalent–divalent degenerate compounds in and around the white area, and to CeN, the only compound that is at the borderline of the tetravalent–trivalent degeneracy. For the Tm chalcogenides, it emerges from XPS measurements that TmS is metallic trivalent, TmSe is intermediate valent and TmTe is semiconducting divalent [48, 49]. This, as can be seen from figure 8, exactly matches the valency sequence that we have derived from the SIC-LSD calculations. For SmS we find $\Delta E_{\text{II} \rightarrow \text{III}} = -5.6$ eV, indicating a divalent semiconductor, but given the small energy difference, susceptible to undergo a localization–delocalization transition under pressure [50], as has been observed experimentally [51]. Apart from EuO, the rare earth monoxides do not appear to occur naturally, but a number of them have been synthesized under high pressure, where it has emerged that EuO and YbO are divalent semiconductors, CeO, PrO and NdO are trivalent metals, while SmO is a metal with the Sm in an intermediate valent state [52]. These findings, including the intermediate valent nature of SmO, are again in excellent agreement with our predictions.

In summary, this effective one-electron SIC-LSD methodology accurately predicts the phase transitions associated with the $f$-electron localization/delocalization. However, we should note here that because of the many-body effects, beyond those included in the SIC exchange-correlation energy functional, the representation of the calculated densities of states in terms of the above-described phase diagram is most likely not the full picture of the underlying electronic structure. This is, in particular, true in the ‘IV’ and heavy fermion areas of the phase diagram, where those ‘left out’ many-body interactions will introduce additional refinements.
into the physical trends based on SIC-LSD. Therefore, compounds that are situated in those areas provide the greater challenge and should be further investigated by methodologies such as the dynamical mean field theory (DMFT) [53] or GW [54].

7. Conclusions and outlook

We have shown that the orbital-dependent DFT approach, based on the LSD energy functional, appended by two important corrections, namely the self-interaction correction and the tetrad effect, can provide a global picture of the electronic structure of the rare earth monopnictides and monochalcogenides, expressed through the uncovered trends in their behaviour. We believe that this SIC-LSD-based study provides an important understanding of the physical properties of these compounds. Although for a given compound the DMFT+LDA or GW approaches can give added insights into the details of the electronic structure, they cannot at the present time be employed for large-scale predictive studies given that they either depend on parameters and/or are very time consuming.

Our study has given rise to a unified picture of the ground state valency and valency transitions across the entire range of rare earth monopnictides and monochalcogenides, starting from tetravalency, through trivalency to divalency, as one moves from the early to late rare earths, in correlation with the ligands. We have also discovered the important links between the underlying electronic structure, valency and physical properties of these compounds, ranging from semiconducting to semi-metallic, metallic and even heavy-fermion-like behaviour.

The present study has also established that if the energy differences between the energetically relevant valency scenarios are small, then fluctuations will play an important role, giving rise to intermediate ground states. This effect can be described within the SIC-LSD formalism by mapping the valence (and also spin) fluctuations on to disorder in the spirit of the Hubbard III approach [55], which we intend to apply in future to the ‘intermediate’ compounds identified in the present study.

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References

[1] Ortiz C, Eriksson O and Klintenberg M 2009 Comput. Mater. Sci. 44 1042
[2] Hohenberg P and Kohn W 1964 Phys. Rev. B 136 864
[3] Kohn W and Sham L J 1965 Phys. Rev. A 140 1133
[4] von Barth U and Hedin L 1972 J. Phys. C: Solid State Phys. 5 1629
[5] Fermi E and Amaldi E 1934 Mem. Accad. Italia 6 119
[6] Temmerman W M, Petit L, Svane A, Szotek Z, Lüders M, Strange P, Staunton J B, Hughes I D and Gyorffy B L 2009 Handbook on the Physics and Chemistry of Rare Earths vol 39 ed K A Gschneidner, J C G Bünzlzi and V K Pecharsky (Amsterdam: North-Holland)
[7] Perdew J P and Zunger A 1981 Phys. Rev. B 23 5048

New Journal of Physics 12 (2010) 113041 (http://www.njp.org/)
[8] Temmerman W M, Svane A, Szotek Z, Winter H and Beiden S V 2000 Lecture Notes in Physics vol 535 ed M Dreysse (Berlin: Springer) pp 286–312
[9] Svane A, Temmerman W M, Szotek Z, Lægsgaard J and Winter H 2000 Int. J. Quantum Chem. 77 799
[10] Svane A 1996 Phys. Rev. B 53 4275
[11] Schmehl A et al 2007 Nat. Mater. 6 882
[12] LeClair P, Ha J K, Swagten J M, Kohlhepp J T, van de Vin C H and de Jonge W J M 2002 Appl. Phys. Lett. 80 625
[13] Duan C G, Sabirianov R F, Mei W N, Dowben P A, Jaswal J S and Tsymbal E Y 2007 J. Phys.: Condens. Matter 19 315220
[14] Petukhov A G, Lambrecht W R L and Segall B 1996 Phys. Rev. B 53 4324
[15] Larson P, Lambrecht W R L, Chantis A and van Schilfgaarde M 2007 Phys. Rev. B 75 045114
[16] Antonov V N, Shpak A P and Yaresko A N 2004 Condens. Matter Phys. 7 211
[17] Ramsden V S et al 2001 Renew. Energy 22 159
[18] Jorgensen C K 1962 Orbitals in Atoms and Molecules (London: Academic)
[19] Nugent L J 1970 J. Inorg. Nucl. Chem. 32 3485
[20] Racah G 1942 Phys. Rev. 62 438
[21] Eriksson O, Brooks M S S and Johansson B 1990 Phys. Rev. B 41 7311
[22] Andersen O K 1975 Phys. Rev. B 12 3060
[23] Tyer R P, Couch P A, van Dam K K, Todorov I T, Bruin R P, White T O H, Walker A M, Austen K F, Dove M T and Blanchard M O 2006 Proc. UK e-Science All Hands Meeting 2006 p 381
[24] Calleja M et al 2004 Proc. UK e-Science All Hands Meeting 2004 p 812
[25] Petit L, Svane A, Szotek Z, Strange P, Winter H and Temmerman W M 2001 J. Phys.: Condens. Matter 13 8697
[26] Johansson B 1979 Phys. Rev. B 20 1315
[27] Patthey F, Bullock E L, Schneider W D and Hulliger F 1993 Z. Phys. B 93 71
[28] Svane A 2006 Solid State Commun. 140 364
[29] Temmerman W M, Szotek Z and Winter H 1993 Phys. Rev. B 47 1184
[30] Temmerman W M, Svane A, Petit L, Lüders M, Strange P and Szotek Z 2007 Phase Transit. 80 415
[31] Villars P and Calvert L D 1991 Pearson’s Handbook of Cystallographic Data for Intermetallic Phases 2nd edn (Ohio: ASM International)
[32] Kershner C J, DeSando R J, Heidelberg R F and Steinmeyer R H 1966 J. Inorg. Nucl. Chem. 28 1581
[33] Johnson D A 2006 Inorganic Chemistry in Focus III ed G Meyer, D Naumann and L Wesemann (Weinheim: Wiley-VCH) p 1
[34] Jayaraman A, Lowe W, Longinotti L D and Bucher E 1976 Phys. Rev. Lett. 36 366
[35] Baer Y, Hauger R, Zührer C, Campagna M and Wertheim G K 1978 Phys. Rev. B 18 4433
[36] Granville S et al 2006 Phys. Rev. B 73 235335
[37] Leuenberger F, Parge A, Felsch W, Fauth K and Hessler M 2005 Phys. Rev. B 72 014427
[38] Aerts C M, Strange P, Horne M, Temmerman W M, Szotek Z and Svane A 2004 Phys. Rev. B 69 045115
[39] Petit L, Svane A, Szotek Z, Temmerman W M and Stocks G M 2010 Phys. Rev. B 81 045108
[40] Steeneken P G, Tjeng L H, Elfmov I, Sawatzky G A, Ghiringhelli D, Brookes N B and Huang D J 2002 Phys. Rev. Lett. 88 047201
[41] Jayaraman A and Maines R G 1979 Phys. Rev. B 19 4154
[42] Ott H R, Rudigier H and Hulliger F 1985 Solid State Commun. 55 113
[43] Monnier R, Degiorgi L and Delly B 1990 Phys. Rev. B 41 573
[44] Wachter P 2001 Physica B 300 105
[45] Horne M, Strange P, Temmerman W M, Szotek Z, Svane A and Winter H 2004 J. Phys.: Condens. Matter 16 5061
[46] Svane A, Santi G, Szotek Z, Temmerman W M, Strange P, Horne M, Vaitheeswaran G, Kanchana V, Petit L and Winter H 2004 Phys. Status Solidi b 241 3185

New Journal of Physics 12 (2010) 113041 (http://www.njp.org/)
[47] Svane A, Strange P, Temmerman W M, Szotek Z, Winter H and Petit L 2001 Phys. Status Solidi b 223 105
[48] Bucher E, Andres K, di Salvo F D, Maita J P, Gossard A C, Cooper A S and Hull G W 1975 Phys. Rev. B 11 500
[49] Kinoshita T, Ufuktepe Y, Nath K G, Kimura S, Kumigashira H, Takahashi T, Matsumura T, Suzuki T, Ogasawara H and Kotani A 1998 J. Electron Spectrosc. Relat. Phenom. 88–91 377
[50] Svane A, Kanchana V, Vaitheeswaran G, Santi G, Temmerman W M, Szotek Z, Strange P and Petit L 2005 Phys. Rev. B 71 045119
[51] Henry D C, Sisson K J, Savage W R, Schweitzer J W and Cater E D 1979 Phys. Rev. B 20 1985
[52] Leger J M, Yacoubi N and Loriers J 1981 J. Solid State Chem. 36 261
[53] Pourovskii L V, Delaney K T, de Walle C G V, Spaldin N A and Georges A 2009 Phys. Rev. Lett. 102 096401
[54] Chantis A N, van Schilfgaarde M and Kotani T 2007 Phys. Rev. B 76 165126
[55] Lüders M, Ernst A, Däne M, Szotek Z, Svane A, Ködderitzsch D, Hergert W, Gyorffy B L and Temmerman W M 2005 Phys. Rev. B 71 205109