Permanently porous hydrogen-bonded frameworks of rod-like thiophenes, selenophenes, and tellurophenes capped with MIDA boronates†

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Permanently porous hydrogen-bonded organic frameworks comprising rod-like molecules with two MIDA boronate termini have been prepared. We show that MIDA boronates self-assemble through multiple hydrogen-bonding interactions. Thiophene-containing frameworks are fluorescent and have a 6.6% absolute quantum yield. The approach appears to be general and introduces new design rules for constructing hydrogen-bonded organic frameworks.

Porous materials such as metal organic frameworks,1–3 covalent organic frameworks (COFs),4–6 and porous aromatic frameworks7,8 have emerged as promising candidates for a range of uses. Porous organic materials include intrinsic and extrinsic porous structures. Organic cage compounds are a typical class of intrinsic porous crystalline solids.9–14 Extrinsic porous materials are formed by the assembly of nonporous small molecules including COFs and the recent revival of hydrogen-bonded organic frameworks (HOFs). HOFs are connected by non-covalent interactions such as hydrogen bonds and various π–π interactions. HOFs that incorporate solvent guest molecules typically lose their porosity after the solvent is removed. As a result relatively few HOFs exhibit permanent porosity.15–22 Permanently porous materials are important for a range of uses,23–26 yet designing building blocks for permanently porous HOFs is challenging.15 The most frequently used building blocks contain the 2,4-diaminotriazinyl group (DAT) and related nitrogen-containing heterocycles (Scheme 1).16 There are a few alternative motifs such as benzimidazolones18 and pyrazoles,19 however, the design rules for HOFs are still in their infancy and very important for the continued development of porous materials.

Herein we report that hydrogen-bonded porous frameworks can be constructed from rod-like molecules that are capped with N-methyliminodiacetic acid (MIDA) boronate termini. The rod-like molecules contain heterocycles that are typical of optoelectronically active materials, including thiophene, selenophene, and tellurophene.27–31 The MIDA boronate group has been applied to a range of cross-coupling synthetic methodologies.32–34 MIDA esters are easily handled, stable under air, and contain both hydrogen bond donors and acceptors. Here we demonstrate that this synthon is useful for the self-assembly of permanently porous materials.

MIDA-containing thiophene, selenophene, and tellurophene rod-like molecules (hereafter denoted DPT-MIDA, DPSe-MIDA, and DPTe-MIDA, respectively) were synthesized by a Pd(PPh₃)₄ and CuI catalyzed Stille-coupling reaction between

Scheme 1 (a) Previously reported functional groups for constructing HOFs. (b) The synthesis of diphenylheterocycle MIDA boronates in this report.
4-iodophenylboronic acid MIDA ester (3.5 eq.) and the corresponding 2,5-bis(trimethylsilyl)heterocycle\textsuperscript{12,16} (1.0 eq.) (Scheme 1). This reaction afforded yellow solids of DPT-MIDA, DPSe-MIDA, and DPTe-MIDA, respectively in high yields (84%–90%). Pure and crystalline products were obtained by recrystallization from acetonitrile. The MIDA boronates were characterized by FT-IR, NMR, and HR-MS (see the ESI\textsuperscript{†}).

The single crystal structures of DPT-MIDA and DPSe-MIDA reveal that two molecules of DPT-MIDA or DPSe-MIDA are contained within each unit cell and the MIDA end groups are in the trans-configuration. DPT-MIDA crystallize through multiple C–H⋯O hydrogen bonding and C–H⋯π interactions (Fig. 1a).

The degree of acetonitrile inclusion is similar in all frameworks. DPSe-MIDA assembled in a similar fashion (see S4, ESI\textsuperscript{†}). Since the only difference is the chalcogen atom, both of the microporous frameworks have a 7 Å × 10 Å pore size along the shortest distance across the channel. Solvent molecules are incorporated in the channel of the frameworks. Three molecules of acetonitrile are incorporated within the DPT-MIDA or DPSe-MIDA unit cells. In DPT-MIDA, one of the acetonitrile molecules is disordered (see S6, ESI\textsuperscript{†}). In both frameworks, guest acetonitrile molecules are bonded with the wall of the channel through weak C–H⋯N hydrogen bonding and C–H⋯π interactions.

DPTe-MIDA was also recrystallized from acetonitrile, however only needle-like crystals could be obtained. In order to compare the structure of DPSe-MIDA with the other samples, powder X-ray diffraction patterns (PXRDs) were collected on all three compounds. For reference the simulated PXRD patterns of DPT-MIDA and DPSe-MIDA were generated from their single crystal data. These closely match the experimental PXRD data (see S8–S9, ESI\textsuperscript{†}), indicating that the structure in the powders are similar to the single crystals. The PXRD scattering pattern of DPSe-MIDA is weak compared to the other two frameworks, which indicates low crystallinity. None-the-less, the main diffraction peaks are close to the other two frameworks (see S10, ESI\textsuperscript{†}), which implies all three frameworks are assembled into a similar structure.

Thermogravimetric analysis (TGA) was performed on each sample under a \textsubscript{N}_2 atmosphere. These experiments reveal that the incorporated acetonitrile molecules are removed at different temperatures. For S and Se, the guest-removal temperatures are around 150 °C; for Te, the temperature drops to 70 °C (see S11–S13, ESI\textsuperscript{†}). The weight loss of DPT-MIDA, DPSe-MIDA, and DPTe-MIDA are 12%, 8.2%, and 6.1% respectively. This is close to the expected weight loss based on acetonitrile inclusion (10%, 9.4%, and 8.7% for DPT-MIDA, DPSe-MIDA, and DPTe-MIDA, respectively). This provides support that the degree of acetonitrile inclusion is similar in all frameworks. No additional weight loss was observed after the loss of acetonitrile until 350 °C, after which the frameworks began to decompose. The incorporation of acetonitrile in the frameworks was also observed in the 1H NMR spectra at δ = 2.07 ppm (see S14–S16, ESI\textsuperscript{†}).

To investigate stability, variable-temperature PXRD patterns were collected on all three frameworks (Fig. 2 and see S20–S21, ESI\textsuperscript{†}). Under air, structural changes were observed. DPT-MIDA is stable up to 100 °C. After that the structure begins to transform, the peaks at 8° and 8.8° become broad, and the peak at 12.2° shifts to 10.7° (Fig. 2). DPSe-MIDA is stable in air up to 125 °C. After that the structure begins to transform, the peaks at 5.4°, 8.2°, and 8.8° become broad, the intensity of peak

\textbf{Fig. 1} (a) Hydrogen-bonding interactions in DPT-MIDA, (b) and (c) microporouos structure of DPT-MIDA and DPSe-MIDA with acetonitrile solvent incorporated.

\textbf{Fig. 2} Variable-temperature PXRD patterns for DPT-MIDA. The asterisks (*) denotes the background signal from a polyether ether ketone layer that protects the detector. The temperature was recorded in 25 °C increments.
10.9° increased, while the peak at 12.3° attenuated (see S20, ESI†). Under vacuum, DPT-MIDA is stable up to 300 °C (see S22, ESI†). DPsSe-MIDA is stable under vacuum as well, even when heated at 150 °C overnight, no obvious change was observed using optical microscopy (see S24, ESI†). Thermally induced transformation are difficult to observe for DPTe-MIDA due to the lower diffraction intensity of this less crystalline sample.

Permanent porosity was evaluated by CO2 adsorption at 273 K. Activated frameworks have 267, 239 and 197 m² g⁻¹ BET surface areas for DPT-MIDA, DPSe-MIDA, and DPTe-MIDA, respectively. The surface area decreases with the increase in molecular weight, which is another sign of similar structure. A slight deviation from linearity is observed which could come from larger atomic radii and lower crystallinity of the Te frameworks (see S25, ESI†). The pore width distributions of DPT-MIDA and DPSe-MIDA have a peak at 5 Å, which is slightly smaller than the pore width expected from single crystal data, likely due to lattice shrinkage after solvent removal (see S26–S27, ESI†). For DPTe-MIDA, two peaks are observed, one at 5.5 Å, and one at 9.8 Å (see S28, ESI†). While the peak at 5.5 Å comes from the ideal crystal structure, the peak at 9.8 Å is almost twice as large and likely come from molecular vacancies in the lattice. DPT-MIDA and DPSe-MIDA take up 39 and 36 cm³ g⁻¹ of CO2 (STP), which corresponds to 1 mole of CO2 adsorbed per mole of building block. DPTe-MIDA takes up 15 cm³ g⁻¹ (STP) of CO2, corresponds to around 0.5 mole of CO2 adsorbed per mole of building block (Fig. 3).

Since thiophene, selenophene, and tellurophene are incorporated into the frameworks, it is interesting to examine their optical properties. The solution absorption maximum (λmax) of the solvated molecules shifts to longer wavelengths (lower energy) as one moves from S (334 nm) to Se (344 nm) to Te (356 nm) in acetonitrile. The emission spectra also red shift from S (412 nm) to Se (424 nm) to Te (439 nm) (see S29–S30, ESI†). The absolute quantum yield of the solid was measured using an integrating sphere.37 Here we chose to focus on DPT-MIDA because it exhibits the strongest fluorescence in solution. The absolute quantum yield of DPT-MIDA is 6.6%. This value is quite high for HOFs and greater than both HOF-111118 and HOF-4,22 which undergo aggregation induced emission. A powder of DPT-MIDA exhibits strong enough blue fluorescence that can be observed with the naked eye (see S31, ESI†).

In conclusion, we have discovered that rod-like molecules that possess two MIDA boronate termini self-assemble into permanently porous HOFs. The thiophene-containing framework exhibits strong photoluminescence in the solid state. Combining rod-like molecules and MIDA esters appears to be a straightforward way to develop porous solids and could be quite general for other types of building blocks with a variety of shapes and symmetry. Thus the design rules presented herein should be useful for new types of frameworks for a range of uses.

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Notes and references

1 M. Eddaoudi, D. B. Moler, H. Li, B. Chen, T. M. Reineke, M. O’ Keeffe and O. M. Yaghi, Acc. Chem. Res., 2001, 34, 319–330.
2 O. M. Yaghi, M. O’ Keeffe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, Nature, 2003, 423, 705–714.
3 M. O’Keeffe, M. A. Peskov, S. J. Ramsden and O. M. Yaghi, Acc. Chem. Res., 2008, 41, 1782–1789.
4 M. Mastalerz, Angew. Chem., Int. Ed., 2008, 47, 445–447.
5 X. Feng, X. Ding and D. Jiang, Chem. Soc. Rev., 2012, 41, 6010–6022.
6 S.-Y. Ding and W. Wang, Chem. Soc. Rev., 2013, 42, 548–568.
7 N. B. McKeown and P. M. Budd, Chem. Soc. Rev., 2006, 35, 675–683.
8 D. Wu, F. Xu, B. Sun, R. Fu, H. He and K. Matyjaszewski, Chem. Rev., 2012, 112, 3959–4015.
9 M. Tozawa, J. T. A. Jones, S. I. Swamy, S. Jiang, D. J. Adams, S. Shakespeare, R. Clowes, D. Bradshaw, T. Hasell, S. Y. Chong, C. Tang, S. Thompson, J. Parker, A. Trewin, J. Bacsa, A. M. Z. Slawin, A. Steiner and A. I. Cooper, Nature, 2009, 8, 973–978.
10 J. T. A. Jones, T. Hasell, X. Wu, J. Bacsa, K. E. Jelfs, M. Schmidtmann, S. Y. Chong, D. J. Adams, A. Trewin, F. Schifffman, F. Cora, B. Slater, A. Steiner, G. M. Day and A. I. Cooper, Nature, 2011, 474, 367–371.
11 T. Hasell, S. Y. Chong, K. E. Jelfs, D. J. Adams and A. I. Cooper, *J. Am. Chem. Soc.*, 2012, **134**, 588–598.

12 M. W. Schneider, I. M. Oppel, A. Griffin and M. Mastalerz, *Angew. Chem., Int. Ed.*, 2013, **52**, 3611–3615.

13 L. Chen, P. S. Reiss, S. Y. Chong, D. Holden, K. E. Jelfs, T. Hasell, M. A. Little, A. Kewley, M. E. Briggs, A. Stephenson, K. M. Thomas, J. A. Armstrong, J. Bell, J. Busto, R. Noel, J. Liu, D. M. Strachan, P. K. Thallapally and A. I. Cooper, *Nat. Mater.*, 2014, **13**, 954–960.

14 G. Zhang, O. Presly, F. White, I. M. Oppel and M. Mastalerz, *Angew. Chem., Int. Ed.*, 2014, **53**, 1516–1520.

15 J.-H. Fournier, T. Maris and J. D. Wuest, *J. Org. Chem.*, 2004, **69**, 1762–1775.

16 Y. He, S. Xiang and B. Chen, *J. Am. Chem. Soc.*, 2011, **133**, 14570–14573.

17 A. I. Cooper, *Angew. Chem., Int. Ed.*, 2012, **51**, 7892–7894.

18 M. Mastalerz and I. M. Oppel, *Angew. Chem., Int. Ed.*, 2012, **51**, 5252–5255.

19 T.-H. Chen, I. Popov, W. Kaveevivitchai, Y.-C. Chuang, Y.-S. Chen, O. Daugulis, A. J. Jacobson and O. Š. Miljanić, *Nat. Commun.*, 2014, 5, DOI: 10.1038/ncomms6131.

20 P. Li, Y. He, H. D. Arman, R. Krishna, H. Wang, L. Weng and B. Chen, *Chem. Commun.*, 2014, **50**, 13081–13084.

21 W. Xiao, C. Hu and M. D. Ward, *J. Am. Chem. Soc.*, 2014, **136**, 14200–14206.

22 H. Wang, B. Li, H. Wu, T.-L. Hu, Z. Yao, W. Zhou, S. Xiang and B. Chen, *J. Am. Chem. Soc.*, 2015, **137**, 9963–9970.

23 Y. Kou, Y. Xu, Z. Guo and D. Jiang, *Angew. Chem., Int. Ed.*, 2011, **50**, 8753–8757.

24 Y. Li, S. Zhang and D. Song, *Angew. Chem., Int. Ed.*, 2013, **52**, 710–713.

25 M. G. Campbell, D. Sheberla, S. F. Liu, T. M. Swager and M. Dinca, *Angew. Chem., Int. Ed.*, 2015, **54**, 4349–4352.

26 P.-F. Li, T. B. Schon and D. S. Seferos, *Angew. Chem., Int. Ed.*, 2015, **54**, 9361–9366.

27 J. Hollinger, A. A. Jahnke, N. Coombs and D. S. Seferos, *J. Am. Chem. Soc.*, 2010, **132**, 8546–8547.

28 A. A. Jahnke, G. W. Howe and D. S. Seferos, *Angew. Chem., Int. Ed.*, 2010, **49**, 10140–10144.

29 L. Li, J. Hollinger, A. A. Jahnke, S. Petrov and D. S. Seferos, *Chem. Sci.*, 2011, 2, 2306–2310.

30 G. L. Gibson, T. M. McCormick and D. S. Seferos, *J. Am. Chem. Soc.*, 2012, **134**, 539–547.

31 Y. S. Park, Q. Wu, C.-Y. Nam and R. B. Grubbs, *Angew. Chem., Int. Ed.*, 2014, **53**, 10691–10695.

32 E. P. Gillis and M. D. Burke, *J. Am. Chem. Soc.*, 2008, **130**, 14084–14085.

33 J. Li, S. G. Ballmer, E. P. Gillis, S. Fujii, M. J. Schmidt, A. M. E. Palazzolo, J. W. Lehmann, G. F. Morehouse and M. D. Burke, *Science*, 2015, **347**, 1221–1226.

34 P. Trinchera, V. B. Corless and A. K. Yudin, *Angew. Chem., Int. Ed.*, 2015, **54**, 9038–9041.

35 D. P. Sweat and C. E. Stephens, *Synthesis*, 2009, 3214–3218.

36 T. Earmme, Y.-J. Hwang, N. M. Murari, S. Subramaniyan and S. A. Jenekhe, *J. Am. Chem. Soc.*, 2013, **135**, 14960–14963.

37 O. Faulkner, J. J. McDowell, A. J. Price, D. D. Perovic, N. P. Kherani and G. A. Ozin, *Laser Photon. Rev.*, 2012, 6, 802–806.

38 Z. Sun, Y. Li, L. Chen, X. Jing and Z. Xie, *Cryst. Growth Des.*, 2015, **15**, 542–545.