Investigation of Adhesion Behaviour of Different Underlayered DLC Coated Cold Forming Tool Steel

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Abstract

Mono- and multilayer coatings can be suitable alternatives to traditional bulk materials due to their favorable mechanical properties. Thanks to the ever-evolving coating technology and process, there are a wide variety of coating types today, such as soft, hard, or super-hard coatings that can be made with single or multilayer coatings.

Adhesion and damage to coatings is especially important for coated tools, where damage to the coating can reduce tool life and result in deterioration of the surface quality of the product. The coatings werecharacterized by Calotest layer thickness measurement, AFM assisted roughness measurement and microhardness test. The adhesion test of the different underlayered (TiBN, CrN and TiAlN) DLC coated Böhler K100 forming tool steel is designed with the Mercedes test (Rockwell C Adhesion Test), which is frequently used to classify the layers, and is complemented by instrumented scratch test.

Author Keywords. Coating, DLC, Tool Steel, Coldforming, Ceramic.

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1. Introduction

The most effective way to increase the tribological performance of sheet metal forming tools is possible with self-lubricating, mono- and multi-layer ceramic coatings (Ghiotti and Bruschi 2011; Podgornik, Hogmark, and Sandberg 2006). Initially, the coatings were applied to the substrate in only one layer –called monolayer coatings – but this did not always form a permanent bond between the coating and the substrate (Carlsson and Olsson 2006). Later, multi-layer coatings appeared, in which not only a single material is applied in the form of an underlayer, but several different layers are formed under the coating.

In recent years, a new generation of coatings has come to the fore. These Diamond-Like-Carbon (DLC) coatings (Tyagi et al. 2019), in addition to their high hardness (1500-3200 HV), are due to their extremely low friction (Nuruzzaman et al. 2011) compared to conventional PVD coatings (Hieke et al. 2005) (e.g. nitride coatings), which can be up to 5-10 times lower (Huang, Guo, and Kao 2010). The unique combination of these properties has opened up new possibilities in the protection against various wear damages, allowing the wear of mechanical parts exposed to wear, such as engine parts, etc. development. These coatings, among other advantages, are generally biocompatible (Allen, Myer, and Rushton 2001; Dearnaley and Arps 2005), therefore, they are also suitable for food processing and medical applications (Ma, Ruys, and Zreiqat 2009), as are PACVD (Mitterer et al. 2003) coatings.

These composite coatings generally comprise carrier layers of different materials, such as Cr, CrN, Si, W and WC-C, followed by an amorphous carbon-enriched upper functional layer.
The selection of the substrate is based on several criteria. These include requirements for adhesion, mode of damage, type and geometry of contact surfaces, applied load and other metallurgical considerations (Pellier et al. 2017).

The DLC coating is a metastable form of amorphous carbon (McColl, Wood, and Grant 1994) formed largely of sp3-hybrid carbon atoms (Celli and Butler 1991).

2. Examination and Results

2.1. Materials

The base material of the test specimens examined during the measurements is K100 cold forming tool steel. The EN number of the tool steel is 1.2080, the EN symbol for its composition is X210Cr12. Its chemical composition is shown in Table 1.

| C   | Si  | Mn  | Cr  | Mo  | Ni  | V   | W   | Co  | Other |
|-----|-----|-----|-----|-----|-----|-----|-----|-----|-------|
| 1,55| 0,30| 0,30| 11,30| -   | -   | -   | -   | -   | -     |

Table 1: Chemical composition of X210Cr12 cold forming tool steel

The fabric structure of the material is ledeburite, there is a slight change in size during heat treatment. The specimens were made of 20.8 mm diameter rods, 20 mm high cylinders by machine cutting in the softened state. The rollers were then ground on the two end faces, followed by heat treatment and polishing. The heat treatment of the K100 cold forming tool steel consists of hardening and tempering. Training is performed in a vacuum oven at 980 °C. Holding was necessary for 15-30 minutes after the full heat treatment. Cooling is possible with
oil, salt bath or gas. In the case of the test specimens, the pieces were cooled in nitrogen under an overpressure pressure of 6 Bar. After hardening, the tempering took place at 350 °C. The hardness after heat treatment on the test pieces was 57-61 HRC.

### 2.2. Thickness and microhardness of coatings

The layer thickness of the coatings was determined using a Calotest equipment. The test was performed with a 30 mm diameter steel ball. Due to differences in the mechanical properties of the coatings, the samples were abraded for different periods of time. TiBN-coated specimens were abraded for 3 min, while for different underlay DLC-coated specimens, a test time of 5 min was used because the upper DLC layer had already ruptured during this abrasion time. The results of the study are summarized in the Table 2.

|                        | TiBN | TiBN+DLC | TiAIN+DLC | CrN+DLC |
|------------------------|------|----------|-----------|---------|
| Thickness of TiBN / TiAIN / CrN layer [µm] | 2,207 | 2,04 | 1,72 | 1,341 |
| Thickness of the WC layer [µm] | - | 0,72 | 0,562 | 0,474 |
| Thickness of DLC layer [µm] | - | 1,461 | 1,472 | 1,5 |
| Σ Thickness [µm] | 2,207 | 4,22 | 3,75 | 3,315 |
| HV0.1 | 401,4 | 474,4 | 475,2 | 552 |

**Table 2:** Thickness and microhardness of multilayered DLC coatings

### 2.3. Surface roughness

An AFM microscope was used to determine the surface roughness of the coatings. A tapping test method was used during the examinations, the size of the area scanned during the examination was 4900 µm. An actual profile was selected during scanning with the touch probe in the test area. During the tests, the 2D roughness parameters of the coated specimens were determined. The results of AFM measurements are shown in Figure 2 and Table 3.

**Figure 2:** AFM surface topography for DLC coated tool steels: TiBN; TiBN+DLC; TiAIN+DLC; CrN+DLC

| SR, mm | TiBN | TiBN+DLC | TiAIN+DLC | CrN+DLC |
|--------|------|----------|-----------|---------|
| Ra     | 156  | 93,4     | 110       | 107     |
| Rz     | 19,3 | 13,9     | 14,2      | 14,3    |

**Table 3:** Ra and Rz roughness parameters for different underlayered DLC coatings

Based on the results, it can be concluded that the use of DLC coatings can significantly reduce the surface roughness. This is especially noticeable for TiBN and TiBN + DLC coatings. However, there is no significant difference in surface roughness for Rz parameters. However, there is a difference in the surface roughness Ra, in which case the TiBN + DLC coating has the lowest roughness and the TiAIN + DLC the largest. Based on these, it can be stated that the effect of the underlayer on the roughness parameter Rz is practically negligible, the roughness parameter Ra can be better applied to characterize these coatings.
2.4. Rockwell C adhesion test

In our research, the adhesion of TiBN, TiBN + DLC, TiAlN + DLC and CrN + DLC coatings was investigated using Rockwell-C adhesion test (Broitman and Hultman 2012). VDI 3198 specifies the Rockwell-C hardness test method (Vidakis, Antoniadis, and Bilalis 2003) for testing coated surfaces (Kayali and Taktak 2015). During the test, a diamond cone was pressed into the specimen with a loading force of 1500 N. The resulting impression was imaged using a light microscope (Heinke et al. 1995). The figures were compared (Haršáni et al. 2018) with the standard illustrations in the standard. The scale (Flores-Renteria et al. 2015) used for visual inspection is illustrated in Figure 3.

Microscopic images of imprints of specimens with different coatings are given in Figure 4. The figure shows, that TiBN, TiAlN + DLC, TiBN + DLC and CrN + DLC coatings are among the acceptable coatings according to the classification in VDI 3198. Despite the defective piercing tool, it can be stated that in the case of TiBN coating no microcracks are visible on the microscopic image, while in the case of TiAlN + DLC coating there is a radial microcrack at the edge of the imprint, but their number is small.

CrN + DLC coatings have smaller cracks than TiAlN + DLC coatings, but are denser along the imprint, however, in the case of CrN-based coatings the layered coating caused by the plastic deformation of the substrate can be observed, but its extent still meets the standard. In the case of the TiBN + DLC coating, the TiBN layer adheres strongly to the base material, however, larger microcracks can already be observed at the edge of the imprint, followed by layered deposition of the coating. This means that the adhesion of the coating to the base material is adequate, however, the coating is brittle due to the brittleness of the DLC coating. However, such damage is considered acceptable as required by the standard.

2.5. Instrumented scratch test

An SP-15 instrumented scratch tester was used to determine the peeling and adhesion properties of the coatings. The custom-developed equipment combines an optical microscope and a special hardness tester. The loading mode was linearly increasing, normal load, the
applied force was 2-150 N, the scratch length is 15 mm, and the feed velocity of table is 5 mm/min. With the set test parameters, one sample for each coating type three scratch tests were performed, the results obtained were the tangential force – friction coefficient diagrams were recorded. Each coating type has friction coefficient and scratch path length relationship for 1-1 scratch test in case of registries. The critical loading force and the coefficient of friction in case of different coatings is shown in Table 4.

| Coating type   | Coefficient of friction, $\mu$ | Critical force, $F_{\text{crit}}$, [N] |
|----------------|---------------------------------|----------------------------------------|
| TiBN           | 0.18                            | 105                                    |
| TiBN+DLC       | 0.17                            | 100                                    |
| TiAlN+DLC      | 0.11                            | 42                                     |
| CrN+DLC        | 0.15                            | 80                                     |

**Table 4:** The friction coefficient and critical force in case of DLC coatings (parameters: linearly increasing, normal loading force: 2-150 N, feed velocity of table: 5 mm/min, scratch length: 10 mm)

Critical loading force ($F_{\text{crit}}$) required to release the coating in the case of a TiBN coating is 105 N and the coefficient of friction is 0.18. Using a DLC layer is critical load value is 0.17, critical load value is 100 N. During the tests it was observed that the coating tears relatively late (10 mm scratch distance), the DLC coated sample had a quieter bouncing sound, which may indicate cracking of the DLC layer (Figure 5).

![Figure 5: Coefficient of friction – scratch length diagram in case of TiBN; TiBN+DLC; TiAlN+DLC and CrN+DLC multilayered coatings](image)

During the scratch inspection of the TiAlN + DLC coating, the coating detached from the TiBN much earlier compared to underlay coatings. In this case, the coating is required for peeling critical load is 42 N, the value of the coefficient of friction compared to TiBN systems lower, around 0.1-0.12.

For CrN + DLC coatings, the coefficient of friction is around 0.15 – similarly for TiBN coating systems – critical load is 80 N. However, it can be seen from the diagram that it takes more time than the coating to tear off in the case of a TiAlN underlayer, however, it detaches earlier than the TiBN coating system. It can be determined on the basis of the scratch tests performed that the various one and the scratch resistance of multilayer coatings is typically different, the layer detachment characteristic critical loading force of the TiBN coated sample is the largest, the DLC in the case of coatings with a top coat, presumably due to early damage to the top coat, a critical load is less.
To investigate the effect of hardness on adhesion properties, the hardnesses of the coatings were plotted as a function of critical loading forces. Based on these, we obtained the following diagram (Figure 6).

![Diagram](image)

**Figure 6:** The relationship between the critical loading force and microhardness of TiBN and multilayered DLC coatings

Based on Figure 6, Equation (1) can be written between hardness and critical loading force:

\[ y = (-0.759x) + 537.8 \]  

(1)

From the above equation and the diagram, it can be concluded that there is a clear relationship between the hardness of the coating and the critical load force, the load force required to detach the coating increases as the hardness of the coating decreases. The main reason for this is that the higher hardness coating is more brittle and therefore bounces off the surface much more easily, while the softer coating is more resistant to plastic deformation caused by scratch needles. This finding can be related to the results of the Mercedes tests of DLC coatings, as there is practically no damage along the imprint of the lowest hardness TiBN coating. At the same time, the higher hardness coatings were less resistant during the test and cracks and detachments were observed along the imprints. However, it is important to note that although the DLC layer may be damaged, the more flexible underlayer underneath can still protect the base material from damage.

3. **Conclusion**

In our research, the adhesion properties of various underlay DLC coatings were investigated using Rockwell adhesion testing and instrumented scratch testing. The results of the studies can be summarized as follows:

1. The Rockwell-C adhesion test is suitable for the qualitative description of the adhesion properties of coatings, but does not provide quantitative results. Based on the tests performed, the coatings can be considered suitable for adhesion.

2. It can be determined based on the scratch tests performed, that the scratch resistance of different single and multilayer coatings is typically different, the critical loading force characteristic of layer deposition in the TiBN coated sample is the largest, the critical loading force is lower for coatings with a DLC topcoat, presumably due to early layer damage.

3. The Rockwell C adhesion test, supplemented by an instrumented scratch test, may be suitable for DLC coated tool steels for a broader description of the adhesion properties of coatings.
4. Studies have shown that the hardness of coatings can affect the scratch resistance of the base material: softer coatings perform better in this respect than coatings with higher hardness.

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Investigation of Adhesion Behaviour of Different Underlayered DLC Coated Cold Forming Tool Steel
Noemi Laszlo

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