Synthesis and characterization of electrically conductive polyethylene-supported graphene films

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Abstract
We describe a simple mechanical approach for low-density polyethylene film coating by multilayer graphene. The technique is based on the exfoliation of nanocrystalline graphite (few-layer graphene) by application of shear stress and allows to obtain thin graphene layers on the plastic substrate. We report on the temperature dependence of electrical resistance behaviors in films of different thickness. The experimental results suggest that the semiconducting behavior observed at low temperature can be described in the framework of the Efros-Shklovskii variable-range-hopping model. The obtained films exhibit good electrical conductivity and transparency in the visible spectral region.

Keywords: Graphene; Shear stress; ITO; Electrical conductor; Transparency; Polyethylene-supported graphene
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Background
The development of techniques for fabricating highly conductive, transparent, and flexible electrodes is the major challenge of organic electronics, and an important topic is the search of alternative materials to replace indium tin oxide (ITO) and fluorine tin oxide (FTO) which are widely used as metal oxide window electrodes in optoelectronic devices. In particular, ITO is practically the only available transparent and electrically conductive material really adequate for industrial applications. However, there are several drawbacks for the use of this material such as, limited availability of indium, poor chemical stability to acid and bases, low near-IR transparency, and easy substrate contamination by ion diffusion. The search for electrode materials with good stability, high transparency, and excellent electrical conductivity is therefore of great importance in optoelectronics. Graphene has emerged as a valuable alternative in view of its high electrical conductivity, flexibility, and good thermal and mechanical stability [1-3]. Recently, transparent and conductive electrodes have been fabricated by including graphene layers in polystyrene or silica but their electrical conductivity was found to be quite low, between 1 and 10−3 S/cm, depending on the amount of graphene [4-8]. An alternative method which involves segregation of the graphene phase, for example, on the surface of a proper substrate allows to obtain high-quality conductive and transparent films.

We describe a simple mechanical technique which makes use of low-density polyethylene film coating by multilayer graphene (PMLG). PMLG is produced by exfoliation of nanographite, i.e., few-layer graphene (FLG), under a combination of shear and friction forces. Since the graphene layers are weakly bound to the surface of FLG [9], they can be easily removed by the action of a shear stress. The mechanism of rubbing a liquid suspension of FLG between two flat and parallel surfaces results in the development of π-π interactions with graphene layers and, ultimately, to complete exfoliation of the nanocrystal in the form of graphene sheets on the substrate surface. If the amount of exfoliated FLG is enough, the surface of the substrate is completely covered by few layers of graphene. Polyolefins, being able to interact with the graphene by CH/π interactions, are ideal substrates for graphene deposition.
by this mechanical procedure. We demonstrate here experimentally that low-density polyethylene substrates are well suited for graphene coating by FLG under the application of shear stress, and we report a systematic investigation of the optical behavior and temperature dependence of electrical transport properties of the PMLG films as a function of the thickness. We show that the dominant transport mechanism in PMLG at low temperatures is due to Coulomb interactions in the hopping regime.

Methods

FLG was prepared according to the following method: expandable-graphite flakes (Faima S.r.l., Milano, Italy) withstand a thermal shock at 750°C for 3 min in a muffle furnace to produce expanded graphite. The expanded graphite filaments were converted to FLGs by ultrasonic treatment in acetone, using a horn sonicator (Bandelin Sonopuls, Mod. UW2200, 20 kHz, 200 W, Berlin, Germany). The suspension (800 ml) was sonicated for 30 min in a glass cylindrical beaker at room temperature (the beaker was placed in a refrigeration bath). The final product was a concentrated colloidal suspension (the FLG concentration in this paste was ca. 33 g/dm³) which was dried in air at room temperature to give the FLG powder. Quite transparent and electrically conductive graphene-coated polymeric films were produced by rubbing out with a low-density polyethylene (LDPE) surface, an alcoholic (ethanol, 99%, Aldrich, St. Louis, MO, USA) suspension of FLG on films of LDPE (20 × 20 cm, with thickness of ca. 80 μm) perfectly adhering to a glass sheet. Then, the FLG excess was removed from the modified LDPE film by washing its surface with pure ethanol. Digital microscopy was used to verify that, at the end of washing process, the films resulted cleaned of residual FLG. The film had several scratches produced by the nanocrystalline graphite (FLG) that possesses hardness slightly higher than LDPE (the hardness in the Shore D scale of LDPE is 55, while for graphite, it is 60 to 80). The film appeared slightly gray colored, and such coloration was uniform over the full treated surface. Film reflectance was depending on the amount of deposited graphene. It was possible to control the graphene coating thickness (average number of deposited layers) by changing the concentration of the alcoholic FLG suspension. A flow diagram of the preparation technique is shown in Figure 1. The optical properties of graphene-coated films were obtained spectroscopically (Lambda-850, PerkinElmer, Waltham, MA, USA). The contribution of LDPE substrate to the optical spectrum was subtracted. The morphological characteristics of the graphene-coated LDPE films were analyzed by atomic force microscopy (AFM), using a microscope (Digital Instruments Nanoscope IIIa, Digital Instruments, Tonawanda, NY, USA) equipped with a sharpened silicon tip having an apical curvature radius of 5 nm. The AFM images were acquired in tapping mode under ambient conditions, with a scan size and rate of 2 μm and 1 Hz, respectively. After performing deconvolution on each AFM image, in order to minimize the tip size effect, the three-dimensional view of the deposits was reconstructed and the root mean square (RMS) roughness was evaluated by means of an image processing technique. Further insight on the film morphology was obtained by scanning electron microscopy (SEM). SEM images were taken with a Jeol JSM-7001 F FEG environmental SEM (Jeol, Akishima, Tokyo, Japan) using a field emission of 10 kV.

The temperature dependence of electrical transport properties was also investigated on PMLG samples of different thickness. The temperature was varied from 50 to 300 K, and the current–voltage characteristic measurements were performed in liquid helium. The electrical contacts consisted in aligned gold pads deposited on the specimen surface by evaporation technique. The samples

![Figure 1 Schematic representation of the shear stress-based method for preparation of polyethylene-supported graphene films.](image-url)
were thermally anchored at the last stage of the probe. The measurement lines are filtered using RC filters with cut-off frequencies of about 1 MHz, thermally anchored at 4.2 K. The sample was voltage-biased with a DC source and the current is measured by a picoammeter.

**Results and discussion**

Atomic force microscopy and scanning electron microscopy images of the samples provide useful information on the film surface structure. AFM image in Figure 2a shows that the coating layer appears to be constituted by graphene polygonal platelets. RMS roughness, obtained from several image analyses, was 10 ± 1 nm on a surface of 2 × 2 \( \mu \)m sized. This peculiar morphology was also observed in higher magnification SEM images (Figure 2b), and it confirms that the mechanical treatment of the PMLG was able to cover the full surface of the LDPE substrate in a continuous and uniform way.

The crystallographic structure of the graphene-based coating was investigated by X-ray powder diffraction (XRD) in \( \theta \)-2\( \theta \) testing mode. The XRD pattern of the FLG precursor (Figure 3a) shows a single peak at \( 2\theta = 26.51^\circ \) which is ascribed to the iso-orientation of graphite nanocrystals corresponding to the (002) graphite peak. The XRD pattern of pure LDPE film (Figure 3b) shows two peaks at a \( 2\theta \) value of 21.3° and 23.6° produced by the crystalline fraction of LDPE and the diffuse halo produced by the amorphous fraction. The XRD in Figure 3c is the diffractogram of the PMLG showing the peaks of the FLG and LDPE. The (002) graphite peak at \( 2\theta = 26.45^\circ \) slightly differs from that of the pure FLG because the crystallographic phase generated during the

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**Figure 2** Typical morphology of PMLG films. (a) AFM image (2 × 2 \( \mu \)m) and (b) SEM image (23 × 18 \( \mu \)m).

**Figure 3** XRD diffractograms of FLG (a), pure LDPE film (b) and LDPE-graphene film (c). The graphene layer thickness, calculated by applying the Scherrer equation to the peak at \( 2\theta = 26.3^\circ \), was 16 nm.
coating process has an increased content of defects (e.g., bending and other damages of the graphene sheet edges induced by the mechanical stress applied to FLG during the coating formation process).

The produced films are optically transparent in the visible range with a thickness ranging from 15 to 30 layers obtained by optical measurements [10]. The transmittance of the PMLG films was measured by UV–vis spectroscopy using a PerkinElmer lambda-850 spectrophotometer.

The transmittance data $T(\%)$ as a function of wavelength for three PMLG films are shown in Figure 4 with the inset showing the relationship between the transmittance at wavelength of 550 nm as a function of the number of layers $N$. The transmittance drops in the observed range following the Beer’s law with an attenuation coefficient close to the theoretical value 2.3%. A transmittance (at 550 nm) of 74% was obtained for a PMLG of 11 graphene layers. The transmittance spectra also exhibit a pronounced absorption band in the UV region at 254 nm due to collective $\pi-\pi^*$ electron transitions.

To further investigate the transport mechanism in PMLG, we studied the temperature dependence of the electrical conductivity in PMLG samples of various number of graphene layers. Figure 5 shows photographs of sample holder used for the PMLG film electrical characterization with a detail of the contact electrodes.

Figure 4: Transmittance of $N$-layer PMLG films. The inset is the relationship between the transmittance, $T(\%)$, at wavelength 550 nm as a function of the number of graphene layers, $N$.

Figure 5: Photographs of the sample holder used for the PMLG film electrical characterization. (a) Sample holder and (b) detail of the electrical contacts.
Figure 6a,b,c shows the representative $I$-$V$ characteristics at different temperatures of three samples of PMLG with $N = 18$, $N = 14$, and $N = 11$ graphene layers, respectively.

The current as a function of voltage $I(V)$ in Figure 6a, b,c shows a finite resistance at zero bias and exhibits a nonlinear behavior with increasing the voltage which is more evident at low temperatures and in sample of lower number of layers. At sufficiently high temperatures, we observe linear $I(V)$ over the considered voltage measurement range. In order to investigate the dominant scattering mechanism, we employed the resistance curve derivative analysis (RCDA) to study the temperature dependence of the conductance [11,12].

Figure 6d compares the temperature dependence of the differential conductance $G_d = dI/dV$ at zero bias ($-0.01 \, V \leq V \leq 0.01 \, V$). The differential conductance of the PMLG samples is plotted versus $T^{1/2}$ on a semilogarithmic scale. We can see that the experimental results are well fitted with the Efros-Shklovskii variable-range-
hopping model (ES-VRH) and suggest that electron–
electron Coulomb interactions are the dominant trans-
port mechanisms and that they are strongly dependent
on temperature. The characteristic temperature de-
pendence of hopping conduction is given by $G(T) = G_0$
$\exp[-(T_0/T)^{1/2}]$ where $G_0$ is a prefactor with $T_0$
being a constant and originates from localized states induced
by charge impurities [13-15].

The red dotted lines in Figure 6d are linear fits of the
conductance data based on the ES-VRH model below
160, 150, and 120 K for PMLG samples of $N = 18$, $N = 14$, and $N = 11$ graphene layers, respectively. With decreasing the number of layers, the PMLG samples become more insulating and we find that deviations from the behavior characteristic of ES-VRH hopping occur at slightly decreasing temperature as can be seen by comparing the temperature dependence of the con-
ductance of the three samples (linear fit from top to bot-
tom in Figure 6d).

Experimental data show that upon increasing the temperature, the conductance increases but the trans-
port cannot be described by ES-VRH over the whole temperature measurement range. In higher temperature
regime, the conductance tends to decrease slightly with increasing temperature, indicating metallic behavior of these films. The sheet resistance $R_s$ of PMLG of 11 gra-
phene layers is as high as about $R_s = 4.5M\Omega/sq$ but it de-
creases significantly with increasing the number of graphene layers. It is about 1.7M$\Omega$/sq in films of 14
layers. The lowest $R_s = 53K\Omega/sq$ with transmittance (at
550 nm) of about 60% was obtained for PMLG of 18 gra-
phene layers. These results demonstrate the fabricated
PMLG films exhibit good electrical conductivity and that there is a trade-off between the conductivity and optical transparency with increasing graphene content.

Conclusions
In summary, we have described a simple mechanical tech-
nique for low-density polyethylene film coating by multi-
layer graphene. This technique is based on the exfoliation
of nanocrystalline graphite by application of shear stress
and allows to obtain thin graphene layers on the plastic
substrate. The temperature dependence of the electrical resistance of PMLG samples of different graphene layers
was investigated. The experimental results suggest that
Coulomb interaction plays an essential role and we showed that Efros-Shklovskii variable-range-hopping is
the dominant transport mechanism at low temperatures.

The advantage of this approach is it is a cheap and simple fabrication procedure. The obtained films exhibit good electrical conductivity and transparency in the visible spectral region which can be of interest for their use as transparent and conductive films alternative to metal ox-
ides in optoelectronic devices.

Competing interests
The authors declare that they have no competing interests.

Authors’ contributions
GC conceived of the experimental design, prepared the samples, and carried out the morphological-structural characterization. SDN participated in the design of the experiment, developed the theoretical analysis, and co-wrote the paper. LN participated in the design of the experiment and coordination. DM performed the electrical and transport measurements from room temperature down to liquid helium temperature. GA carried out the samples’ morphological characterization and analysis by means of atomic force microscopy. GPP participated in the design and coordination of the experimental measurements. All authors read and approved the final manuscript.

Authors’ information
GC is a senior researcher of the Italian National Research Council, Institute for Polymers, Composites and Biomaterials. His present research interests are in the field of advanced functional materials based on polymer-
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Based materials and techniques for the graphene chemical modification in addition to morphological, structural, and spectroscopic characterization methods are studied. He has authored 150 research articles published in international journals, ten patents, and many conference papers. He is the editor of two Wiley books devoted to metal-polymer nanocomposites and is a member of the editorial board of different scientific journals.

SDN got the degree in physics (1082) at ‘Federico II’ University of Naples, Italy. From 1983 to 1987, he was a system analyst at Eletronica S.p.A. (Rome) and Aenia S.p.A. (Naples). Since 1988, he has been a staff researcher at the Institute of Cybernetics ‘E. Caianiello’ of the National Research Council (CNR). Currently, he is a senior researcher of the Italian National Research Council, Institute for Superconductors, Oxide Materials and Devices. He has been a scientific coordinator of the research project ‘Imaging Techniques for Studying and Analyzing Microstructured Materials’ of the Department of Physics Sciences and Matter Technologies (DSFTM) of the National Research Council. He has authored about 300 research articles in peer-reviewed international journals, book chapters, and conference proceedings and 7 patents. He has served in program committees of several international conferences and has been a referee for various journals in the field of optics and theoretical physics. His research interests include the development of quantum methodologies to the description of coherent phenomena in many body systems, quantum tomography, theoretical modeling for studying dynamical effects in mesoscopic systems and nanostructured polymeric materials, electronic coherent transport in unconventional superconductors and graphene, and interaction of optical and electron beams in nonlinear media and plasma.

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GA got the degree in Physics (1997) and PhD degree in Materials Engineering (2004) at ‘Federico II’ University of Naples, Italy. He is currently a researcher at ‘Federico II’ University of Naples, Italy. His main research topics
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