Pressure-induced phase transitions and superconductivity in magnesium carbides

Sooran Kim1,2, Kyoo Kim1,3,7, Jahyun Koo4, Hoonkyung Lee4, Byung Il Min1 & Duck Young Kim5,6*

Crystal structure prediction and in silico physical property observations guide experimental synthesis in high-pressure research. Here, we used magnesium carbides as a representative example of computational high-pressure studies. We predicted various compositions of Mg–C compounds up to 150 GPa and successfully reproduced previous experimental results. Interestingly, our proposed MgC2 at high pressure >7 GPa consists of extended carbon bonds, one-dimensional graphene layers, and Mg atomic layers, which provides a good platform to study superconductivity of metal intercalated graphene nano-ribbons. We found that this new phase of MgC2 could be recovered to ambient pressure and exhibited a strong electron-phonon coupling (EPC) strength of 0.6 whose corresponding superconductivity transition temperature reached 15 K. The EPC originated from the cooperation of the out-of-plane and the in-plane phonon modes. The geometry confinement and the hybridization between the Mg s and C p orbitals significantly affect the coupling of phonon modes and electrons. These results show the importance of the high-pressure route to the synthesis of novel functional materials, which can promote the search for new phases of carbon-based superconductors.

Carbon has attracted much attention as a key element of contemporary science and technology. It is even considered as the most promising platform for emergent energy materials, which might be able to replace the current main player, silicon, in the near future. Indeed, during the last few decades, this perspective has been realized through the experimental synthesis of novel low-dimensional carbon systems such as carbon nanotubes, fullerenes, and graphenes. Unlike other carbon-group elements, such as Si and Ge that possess only sp3 bonding at ambient conditions, carbon allows different orders of bonding such as sp2 and sp3 as well as sp1, which makes the phase diagram of carbon very complicated and can eventually lead to various interesting structures.

Furthermore, carbon compounds showing superconductivity have been intensively and extensively studied. Hannay et al., first reported graphite intercalation compound superconductors (GICs) with the superconducting critical temperature (Tc) < 1 K (C8A, A = K, Rb, Cs). The recent discovery of YbC6 (Tc = 6.5 K), CaC6 (Tc = 11.5 K) promoted an interest in GICs, and various GICs have been since reported. CaC6 produces the highest Tc = 15.1 K in GICs under the pressure of 7.5 GPa. A graphene-nanoribbon structure with Ca, CaC2, was predicted to exhibit superconductivity with a Tc = 9.8 K at 95 GPa.

It is believed that the phonon mediates the superconductivity in the metal-intercalated carbon superconductors, and the electron-phonon coupling (EPC) is the underlying mechanism of their superconductivity. Therefore, analyzing the phonon modes is an essential step to explore the mechanism of the superconductivity. Roughly, two important vibration modes exist: One is a high-frequency in-plane mode, and another is a low-frequency out-of-plane mode. The EPC constant \( \lambda = N(0)D^2/M\omega_{ph}^2 \) can be enhanced by the large D and low energetic \( \omega_{ph} \), where N(0) is the density of states (DOS) at Fermi level (EF), D is the deformation potential that is related to coupling between the electron and phonon mode, M is the effective atomic mass, and \( \omega_{ph} \) is the phonon frequency. Despite the large D of the in-plane modes, \( \lambda \) is small because of the highly energetic \( \omega_{ph} \) of the in-plane modes.

1Department of Physics, Pohang University of Science and Technology, Pohang, 37673, South Korea. 2Department of Physics Education, Kyungpook National University, Daegu, 41566, Korea. 3Max Plank–POSTECH/Hsinchu Center for Complex Phase Materials, Pohang University of Science and Technology, Pohang, 37673, South Korea. 4Department of Physics, Konkuk University, Seoul, 05029, South Korea. 5Center for High Pressure Science and Technology Advanced Research, Shanghai, 201203, China. 6Department of Chemistry, Pohang University of Science and Technology, Pohang, 37673, South Korea. 7Present address: Korea Atomic Energy Research Institute (KAERI), 111 Daedeok-daero 989 Beon-Gil, Yuseong-gu, Daejeon, 34057, South Korea. *email: duckyoung.kim@hpstar.ac.cn
C contains a C4 chain, which is more stable than the experimentally known phase at low pressure. Interestingly, this structure has dynamic stability not only at high pressure but also at ambient pressure (See Fig. 3a), implying that it is a meta-stable structure of MgC2. The dumbbell and pentagon structures are both an insulating phase22,23. Using first principles calculations24, the calculated maximum $T_c$ of the pentagon structure using $\text{Doped-picene}$, which is a hydrocarbon superconductor with $T_c$ of 15 K17, is reported to show that intramolecular in-plane carbon vibrations can contribute to the enhanced electron-phonon coupling18.

Mg–C compounds could be promising superconducting materials because various carbon-bonding motifs can exist in magnesium carbides19. Recently, new phases of MgC and $\beta$-MgC2 were reported by experimental and theoretical cooperation19,20. MgC2 crystallizes in a tetragonal structure with the space group $P4_{2}/mnm$. It contains a C2 dumbbell structure, similar to that of CaC221. Srepusharawoot et al. proposed a new phase of MgC2 containing the pentagon structure using $ab$ initio random searching22. The pentagon structure is energetically more stable than the dumbbell structure according to their calculations, which implies that the dumbbell structure could be a meta-stable structure of MgC2. The dumbbell and pentagon structures are both an insulating phase22,23. Furthermore, Wang et al., reported the structural evolution and the superconductivity of MgC2 under pressure using first principles calculations24. The calculated maximum $T_c$ for orthorhombic $Cnmc$ and monoclinic $C2/m$ structures are 11.3 K at 4 GPa and 7.1 K at 9.6 GPa, respectively. Therefore, it is worth studying the possibility of superconductivity at ambient pressure and the origin of the high $T_c$ further.

In this paper, we systematically study Mg–C compounds under high-pressure using first-principles calculations. First, we perform crystal structure searching to find unprecedented compounds, and to verify existing experimentally-known compounds. Then, we investigate phases of MgC2 under pressure up to 150 GPa. The metallic phase of MgC2 is obtained and stable even at ambient pressure, which exhibits superconductivity with a $T_c$ of 15 K. The large $\lambda$ originates from the cooperation of both couplings of the out-of-plane and in-plane modes to charge carriers. Geometry confinement plays an essential role in enhancing the in-plane vibration contribution to the EPC.

**Results**

**Crystal structure searching under pressure.** Figure 1 shows computational structure-searching results of Mg–C compounds at various pressures. At ambient pressure in Fig. 1a, the compounds are energetically unstable while two experimentally meta–stable compounds of MgC2 and MgC3 are close to the stability line (horizontal dashed lines at zero). At 15 GPa, MgC is stable with an anti–fluorite type structure, which has been reported experimentally20. Kurakevych et al. stated that their synthesis condition was 15 GPa20. This is in excellent agreement with our computational prediction. As the pressure increases, the formation enthalpy of MgC2 decreases from $-0.3 \text{eV/atom}$ at ambient pressure to $-0.11 \text{eV/atom}$ at 15 GPa. At a high pressure of 150 GPa, MgC2 can be one of the stable stoichiometries, as shown in Fig. 1b. This stable structure at high pressure will be discussed later. The enhanced stability of Mg–C compounds under high pressure is consistent with a previous report25.

Figure 2a shows the formation enthalpy plot of MgC2 with a reference line exhibiting the previously observed dumbbell structure in the space group $P4_{2}/mnm$. It is worth noting that some of our predicted phases show the lower enthalpy than that of the experimental structure. At ambient pressure, the Mg-intercalated graphene compound ($Pmn2_1$) is predicted to be the ground state. However, when we tested its dynamical stability using $ab$ initio lattice dynamics, it exhibits an imaginary acoustic phonon. Thus, it is unstable against decomposition. As it is a combination of graphene and Mg layers, it would be dissociated into individual graphene and Mg, which is manifested by the negligible charge transfer between C and Mg. Under pressure, we predict two structures; one is $a-I4/mmm$ at low pressure $<7 \text{GPa}$, and another is $C2/m$ structure at high pressure $>7 \text{GPa}$. The $a-I4/mmm$ phase contains a C4 chain, which is more stable than the experimentally known phase at low pressure. Interestingly, $C2/m$ consists of a single zigzag graphene nano-ribbon (zGNR) and Mg array. Hereafter, we refer to this $C2/m$ phase as MgGNR. Figure 2b,c illustrates the top view and side view of the primitive cell of the $C2/m$ structure. This structure has dynamic stability not only at high pressure but also at ambient pressure (See Fig. 3a), implying...
that the MgGNR phase can exist as a meta-stable structure at ambient pressure. The distances of C1-C2 and C2-C2 at ambient pressure are 1.53 Å and 1.45 Å, respectively.

Electron-phonon coupling and superconductivity. Figure 3 shows the phonon dispersion curve and EPC properties of MgGNR at ambient pressure. The phonon dispersion curve and total phonon DOS without phonon softening in Fig. 3a indicates the dynamic stability of the structure. This phase is theoretically predicted to exhibit superconductivity. The superconducting parameters have been calculated using the Eliashberg EPC theory and the Allen-Dynes formula:

\[
T_c = \frac{\omega_{\log}^2}{1.2} \exp \left[ \frac{-1.04(1 + \lambda)}{\lambda(1 - 0.62\mu^*) - \mu^*} \right] + \frac{1}{\omega_{\log}^2} \int \frac{\omega}{2\lambda} F(\omega) \log \omega \, d\omega
\]

where \( \alpha^2 F(\omega) \) is the Eliashberg function, \( \omega_{\log} = \exp \left[ \frac{1}{\lambda} \int \frac{\omega}{2\lambda} F(\omega) \log \omega \, d\omega \right] \) and \( \mu^* \) is the effective Coulomb repulsion parameter. The predicted \( T_c \) of 15 K in Table 1 is close to the maximum \( T_c \) of 15.1 K of existing GICs.

To explain this high \( T_c \), we analyzed the \( \lambda_{\nu} \) and corresponding phonon modes. Interestingly, two regions give a large contribution to the total \( \lambda \) and Eliashberg function, as shown in Fig. 3a, b. The first region is around 30–60 meV at relatively low frequency, and the second one is around 140 meV. Two representative phonon modes at the \( \Gamma \) point produce large contribution to \( \lambda \). The frequencies of the first mode, \( \nu_1 \), and the second mode, \( \nu_2 \), are 45.8 meV with \( \lambda_{\nu_1} = 0.53 \) and 142.9 meV with \( \lambda_{\nu_2} = 0.36 \), respectively. Figure 3c represents the lattice displacements by \( \nu_1 \), which is the out-of-plane vibration mode by Mg and carbon displacements. This is consistent with existing GICs whose out-of-plane mode is usually chosen to explain their superconductivity. The in-plane vibration mode, \( \nu_2 \), corresponds to the in-plane carbon displacement mode, as shown in Fig. 3d. The in-plane phonon mode is very similar to the phonon mode that mediates the superconductivity in MgB2. Interestingly, MgGNR exhibits superconductivity induced by not only the out-of-plane mode but also the in-plane mode. These multiple phonon contributions to \( \lambda \) can produce the high \( T_c \) in MgGNR.

Electronic structures without/with phonon modulation. To investigate the properties of the predicted MgC2 and how \( \nu_1 \) and \( \nu_2 \) contribute to superconductivity, we performed electronic structure calculations without/with phonon modulations. The band structure of MgGNR in Fig. 4a shows the metallic property of the material. The \( p_z \) orbitals of C1 and C2, and the Mg \( s \) orbital are mainly observed near \( E_F \). The other \( p_x \) and \( p_y \) orbitals of C1 and C2 do not contribute near \( E_F \). Each Mg \( p \) orbital contributes less than the Mg \( s \) orbital. The crossing Mg \( s \) orbital corresponding to the interlayer state at \( E_F \) is consistent with that of superconducting GICs, which is proposed as a necessary condition for superconductivity. Both the C \( p_z \) and Mg \( s \) bands are dispersive and show a strong hybridization feature except along the K2-K3 path. The \( p_z \) orbital of C2 is dominant and forms the
flat band along the K2-K3 near EF. The flat band by C2 pz shows the two-dimensional (2D) feature near EF. This electron-like pz state along K2-K3 resembles the hole-like σ state along Γ-A in MgB2, making a cylinder-shape Fermi surface from the 2D nature. Figure 4b represents the Fermi surface consisting of two sheets. The green sheet is derived from hybridization of the C pz and Mg s orbitals, while the blue one is derived from the pz orbitals of C2. Both Fermi surfaces are electron pockets. The blue Fermi surface by C2 pz has a flattened cylinder-shape indicating its 2D character. The similarity to GICs and MgB2 suggests two different charge carriers; one is the electron in the Mg Fermi surface in green coupled with the out-of-plane phonon mode, and another is the electron in C Fermi surface in blue coupled with the in-plane phonon mode. This feature in the band structure calculations with deformed structures by each phonon mode will be discussed later. It is worth noting that the two-band superconductivity from the interband anisotropy is known to be important for MgB2. Compared to the isotropic calculation, considering the interband anisotropy increases the calculated electron-phonon coupling constant and predicted Tc from 0.77 and 22 K to 1.01 and 40 K, respectively. Based on the similarity of MgC2 to MgB2, it would be interesting to investigate the anisotropic effect in MgC2 as future work.

To investigate the role of the Mg in the system, we set the hypothetical structure (HS) by removing the Mg atom from the MgGNR structure. The flat band along K2-K3 of HS in Fig. 4c is almost intact compared to that of MgGNR. The deformation of the bands by Mg orbitals mainly occurs in unoccupied pz bands along Γ-K1-K2 and K3-Γ-K4. Another significant difference between MgGNR and HS is the position of EF. The band structure in HS is shifted upward by ~4 eV with respect to that in MgGNR. MgGNR has the electron in the C2 pz band whereas HS

| (states/eV/f.u.) | ω_2g(K) | Θ_D(K) | λ | Tc(K) | μ = 0.1 |
|------------------|---------|--------|---|-------|---------|
| 0.40             | 594.18  | 717.96 | 0.62 | 14.9  |         |
has the hole in the C2 $p_z$ band, indicating that the role of Mg is electron doping to unoccupied $p_z$ bands of GNR, forming the interlayer state at $E_F$.

Figure 4d shows the charge density plot near $E_F$. Note that $p_z$ bands of C2 in zGNR are strongly deformed compared to those of graphite. The geometry confinement produces the bonding state between the $p_z$ orbitals of two C2s. The top view of charge density resembles the $\sigma$ bonding observed in MgB$_2$. This deformation of the orbital shape can enhance the coupling of the in-plane phonon mode to the C2 $p_z$ orbital. Therefore, the bonding state between the $p_z$ orbitals of C2s increases the deformation potential $D$ of the in-plane phonon mode, and provides the sizable $\lambda$ despite its high frequency. This result also explains the difference and similarity between MgGNR and MgB$_2$. Because MgGNR has two more electron per f.u. than MgB$_2$, the C $p_z$ orbital is mainly located at $E_F$ whereas the B $p_{x,y}$ orbitals are at $E_F$ in MgB$_2$.

On the other hand, the in-plane phonon mode produces a larger deformation in the whole Brillouin zone (BZ) as in Fig. 5e–h. Both green and blue Fermi surfaces are modified by the in-plane phonon mode. The blue Fermi surface by C2 $p_z$ shows a huge modification in contrast to the case of the out-of-plane mode. It qualitatively shows that the deformation potential, $D$, of the in-plane mode is larger than that of the out-of-plane mode. The C2 $p_z$ electron-like state also can be the charge carrier coupled with the in-plane phonon mode. Therefore, our results demonstrate that despite the high $\omega_{ph}$ of the in-plane vibration mode, it can contribute to the EPC with a large enough $D$.

In addition, we checked the HS structure (where Mg atoms are removed) with respect to the atomic displacements of each phonon mode. The electronic structures of the distorted structures are almost intact with respect to the out-of-plane phonon mode, while the electronic structures of the deformed structures by the in-plane mode are modified even without Mg states. This clearly suggests that the coupling between the electrons and the
Properties of MgC$_2$ at 150 GPa. MgC$_2$ undergoes a structural transformation to an ionic compound at around 150 GPa and is energetically stable, as predicted in Fig. 1b. The high-pressure phase of MgC$_2$ possesses an intriguing geometry consisting of two-dimensional magnesium layer sandwiched by two buckled graphene layers, as shown in Fig. 6a,b, in good agreement with a previous theoretical prediction$^{24}$. The carbon atoms consist of out-of-plane mode needs the interlayer state, whereas the coupling between the C$_p$ orbital and in-plane mode is not significantly affected by the interlayer state.

Figure 5. Band structures and Fermi surfaces of distorted structures with respect to the phonon modes. (a–d) the out-of-plane mode, $q_1$, and (e–h) the in-plane mode, $q_2$. Each mode has two band structures and two Fermi surfaces because of the two opposite directions of the corresponding displacements. The structures in (a,c,e,g) with the corresponding phonon mode in the positive direction while the structures in (b,d,f,h) with the corresponding phonon mode in the negative direction. The inset figures in (e,f) show the distorted cGNR by $q_2$.

Figure 6. A high-pressure phase of MgC$_2$. The crystal structure of MgC$_2$ at 150 GPa with solid lines showing the unit-cell. (a) Top view and (b) side view. (c) Band structure of MgC$_2$ and (d) stable phonon dispersion curve at 150 GPa. We chose the continuous $\Gamma$-M-K-$\Gamma$-A-L-H-A path followed by discontinuous M-L and K-H paths.
A hexagonal honeycomb lattice whose lattice constant of 2.42 Å (C–C distance, 1.48 Å) is comparable to the lattice constant 2.46 Å for graphene. The structure is similar to the BCS-type high-$T_c$ superconductor, MgB$_2$, except that the carbon atoms are not located on a plane due to a distortion along the $z$ direction. It is a semiconductor with indirect–bandgap of 0.42 eV, as shown in Fig. 6c. The phonon dispersion curve without imaginary bands indicates its dynamical stability (See Fig. 6d).

Figure 7a,b show the band structure and phonon bands of the high-pressure phase at ambient pressure, respectively. The phonon dispersion curve remains stable down to ambient pressure as in Fig. 7b, which implies its meta-stability at ambient conditions. Note that the band structure at ambient pressure in Fig. 7a shows that the conduction band minimum and valence band maximum become close to the $E_F$ near $A$ point. We further engineered the band gap by adjusting lattice parameters. A closer look at the bandgap in Fig. 7c exhibits a small gap opening at the $A$ point with 1% expansion of $a$ and $c$ lattice parameters. The band gap is closed with 5% lattice expansion in Fig. 7d.

**Discussion**

We computationally studied the high-pressure phases of MgC$_2$. The phases show dynamic stability not only at high pressure but also at ambient pressure. From an application perspective, this feature is intriguing because the properties of this newly-designed materials can be exploited both at high and ambient pressures. The superconductivity of MgC$_2$ is predicted to have a $T_c$ of 15 K, which is high among similar carbon-based superconductors. The $T_c$ originates from the cooperation of the out-of-plane and in-plane phonon modes. The hybridization between Mg $s$ and C $p_z$ produces an interlayer state at $E_F$, and the coupling with the out-of-plane mode. The geometry confinement deformed the C $p_z$ orbital to the bonding state of C $p_z$ orbitals, which is similar to the $\sigma$-bonding states in MgB$_2$, and produces the large deformation potential of the in-plane mode. The graphene-nanoribbon superconductor, MgC$_2$, shows the coupling mechanism between charge carriers and phonon modes observed in GICs and MgB$_2$. Thus, we believe that metal-intercalated graphene-nanoribbon superconductors like MgC$_2$ can be a model system to investigate the relationship between geometry confinement and superconductivity in low-dimensional systems.
Methods

To investigate ground state structures of various Mg-C compounds containing up to four formula units under pressure, a structure searching algorithm USPEX \(^{25,26}\) combined with VASP was employed. \(^{34}\) We conducted USPEX calculations up to 12 atoms in the simulation cell and set the maximum generation number to 50 and the structure number to 60. The calculations were converged within the limit. The generalized gradient approximation of Perdew, Burke, and Ernzerhof (PBE) was used to describe the exchange correlation functional. \(^{35}\) PAW pseudopotential \(^{36}\) was used in the USPEX-VASP calculations with energy cutoff 500 eV and \(2 \times 2 \times 0.01 \text{Å}^{-1}\) Brillouin zone grid of spacing to show a converged energy. After obtaining the structure at high pressure, the structure at ambient pressure was generated by the full relaxation of the atomic positions and the volume.

Harmonic phonon and Electron phonon coupling (EPC) calculations were conducted using Quantum Espresso software \(^{27}\) based on density functional perturbation theory. \(^{38}\) We utilized ultrasoft pseudopotential and PBE functional for EPC calculations. EPC matrix elements were computed in the first BZ on a \(6 \times 6 \times 6\) q-mesh with a \(24 \times 24 \times 24\) k-mesh and \(60\) Ry energy cutoff. The convergence of EPC matrix elements was checked with a \(12 \times 12 \times 12\) q-mesh with a \(24 \times 24 \times 24\) k-mesh. To evaluate the integrated EPC constant (\(\lambda\)), we used the Gaussian broadening of 0.025 Ry.

The band structure calculations were performed with the full-potential linearized augmented plane wave band method implemented in Wien2k package. \(^{29}\) Generalized gradient approximation (GGA-PBE) is used as the exchange-correlation.

Received: 26 July 2019; Accepted: 12 December 2019;
Published online: 27 December 2019

References

1. Hannay, N. B. et al. Superconductivity in graphitic compounds. Phys. Rev. Lett. 14, 225–226 (1965).
2. Weller, T. E., Ellerby, M., Saxena, S. S., Smith, R. P. & Skipper, N. T. Superconductivity in the intercalated graphite compounds C\(_6\)Yb and C\(_6\)Ca. Nat. Phys. 1, 39–41 (2005).
3. Emery, N. et al. Superconductivity of bulk CaC\(_6\). Phys. Rev. Lett. 95, 087003 (2005).
4. Dresselhaus, M. S. & Dresselhaus, G. Intercalation compounds of graphite. Adv. Phys. 51, 1–186 (2002).
5. Heguri, S. et al. Superconductivity in the graphite intercalation compound BeC\(_2\). Phys. Rev. Lett. 114, 247201 (2015).
6. Smith, R. P. et al. Superconductivity in graphite intercalation compounds. Phys. C: Supercond. its Appl. 514, 50–58 (2015).
7. Kubozone, Y. et al. Recent progress on carbon-based superconductors. J. Phys.: Condens. Matter 28, 334001 (2016).
8. Gauzzi, A. et al. Enhancement of superconductivity and evidence of structural instability in intercalated graphite CaC\(_6\) under high pressure. Phys. Rev. Lett. 98, 067002 (2007).
9. Li, Y.-L. et al. Pressure-induced superconductivity in CaC\(_6\). Proc. Natl. Acad. Sci. 110, 9289–9294 (2013).
10. Profeta, G., Calandra, M. & Mauri, F. Phonon-mediated superconductivity in graphene by lithium deposition. Nat. Phys. 8, 131–134 (2012).
11. Calandra, M. & Mauri, F. Theoretical explanation of superconductivity in C\(_6\)Ca. Phys. Rev. Lett. 95, 237002 (2005).
12. Mazin, I. I. Intercalant-driven superconductivity in YbC\(_6\) and CaC\(_6\). Phys. Rev. Lett. 95, 227001 (2005).
13. Csányi, G., Littlewood, P. B., Nevidomskyy, A. H., Pickard, C. J. & Simons, B. D. The role of the interlayer state in the electronic structure of superconducting graphite intercalated compounds. Nat. Phys. 1, 42–45 (2005).
14. Calandra, M. & Mauri, F. Possibility of superconductivity in graphite intercalated with alkaline earths investigated with density functional theory. Phys. Rev. B 74, 094507 (2006).
15. Naganatsu, J., Nakagawa, N., Muranaka, T., Zenitani, Y. & Akimitsu, J. Superconducting at 39 k in magnesium diboride. Nat. 410, 63–64 (2001).
16. Choi, H. J., Roundy, D., Sun, H., Cohen, M. L. & Louie, S. G. The origin of the anomalous superconducting properties of MgB\(_2\). Nat. 418, 758–760 (2002).
17. Mitsuhashi, R. et al. Superconductivity in alkali-metal-doped picene. Nat. 464, 76–79 (2010).
18. Subedi, A. & Boeri, L. Vibrational spectrum and electron-phonon coupling of doped solid picene from first principles. Phys. Rev. B 84, 020508 (2011).
19. Strobel, T. A. et al. Synthesis of MgC\(_3\): A monoclinic high-pressure polymorph of magnesium sesquicarbide. Inorg. Chem. 53, 7020–7027 (2014).
20. Kurakevych, O. O., Strobel, T. A., Kim, D. Y. & Cody, G. D. Synthesis of MgC\(_6\): A magnesium methaneide. Angewandte Chemie Int. Ed. 52, 8930–8933 (2013).
21. Srepusharawoot, P., Blomqvist, A., Araújo, C. M., Scheicher, R. H. & Ahuja, R. One-dimensional polymeric carbon structure based on five-membered rings in alkaline earth metal dicarbides BeC\(_2\) and MgC\(_2\). Phys. Rev. B 82, 125439 (2010).
22. Xiao, B., Feng, J., Chen, J. & Yu, L. Crystal structures and electronic properties of MC\(_2\) (M=Mg;Ca;Sr;Ba) by comparative studies based on ab-initio calculations. Chem. Phys. Lett. 448, 35–40 (2007).
23. Wang, D., Yan, Y., Zhou, D. & Liu, Y. Evolution of crystal and electronic structures of magnesium dicarbide at high pressure. Sci. Reports 5, 17815 (2015).
24. Liu, H., Gao, G., Li, Y., Hao, J. & Tse, J. S. Crystal structures and chemical bonding of magnesium carbide at high pressure. The J. Phys. Chem. C 119, 23168–23174 (2015).
25. Elaisshberg, G. Interactions between electrons and lattice vibrations in a superconductor. Sov. Phys. - JETP (Engl. Transl.); (United States) 11 (1960).
26. Allen, P. B. & Dynes, R. C. Transition temperature of strong-coupled superconductors reanalyzed. Phys. Rev. B 12, 905–922 (1975).
27. An, J. M. & Pickett, W. E. Superconductivity of MgB\(_2\): Covalent bonds driven metallic. Phys. Rev. Lett. 86, 4366–4369 (2001).
28. Kortus, J., Mazin, I. I., Belashchenko, K. D., Antropov, V. P. & Boyer, L. S. Superconductivity of metallic boron in MgB\(_2\). Phys. Rev. Lett. 86, 4656–4659 (2001).
29. Liu, A. Y., Mazin, I. I. & Kortus, J. Beyond elaisshberg superconductivity in MgB\(_2\): Anharmonicity, two-phonon scattering, and multiple gaps. Phys. Rev. Lett. 87, 087005 (2001).
30. Iavarone, M. et al. Two-band superconductivity in MgB\(_2\). Phys. Rev. Lett. 89, 187002 (2002).
31. Oganov, A. R. & Glass, C. W. Crystal structure prediction using ab initio evolutionary techniques: Principles and applications. The J. Chem. Phys. 124, 244704 (2006).
32. Glass, C. W., Oganov, A. R. & Hansen, N. Uspex—evolutionary crystal structure prediction. Comput. Phys. Commun. 175, 713–720 (2006).
33. Kresse, G. & Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. Phys. Rev. B 54, 11169–11186 (1996).
35. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. Phys. Rev. Lett. 77, 3865–3868 (1996).
36. Blöchl, P. E. Projector augmented-wave method. Phys. Rev. B 50, 17953–17979 (1994).
37. Giannozzi, P. et al. QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials. J. Physics: Condens. Matter 21, 395502 (2009).
38. Baroni, S., de Gironcoli, S., Dal Corso, A. & Giannozzi, P. Phonons and related crystal properties from density-functional perturbation theory. Rev. Mod. Phys. 73, 515–562 (2001).
39. Blaha, P., Schwarz, K., Madsen, G. K. H., Kvasnicka, D. & Luitz, J. Wien2k, An augmented plane wave and local orbitals program for calculating crystal properties (2001).

Acknowledgements
This work was supported by NSFC of China (Grant 11774015) and NRF of South Korea (NRF-2017R1D1A1B03031913). K.K. was supported by NRF through Basic Research Programs (No. 2016R1A1B02008461 and no. NRF-2017M2A2A6A01071297), and MPK (No. 2016K1A4A4A01922028). H.L. acknowledges the support from KISTI under the Supercomputing Applications Support Program (KSC-2014-C3-031). B.I.M. was supported by NRF (No. 2017R1A2B4005175). S. K. acknowledges support from the National Research Foundation of Korea (No. 2019R1F1A1052026).

Author contributions
D.Y.K. supervised the research and performed the structure searching under pressure, and electron-phonon coupling calculations. S.K., K.K. and B.I.M. calculated and analyzed the superconductivity and electronic structure data. J.K. and H.L. conducted the calculations of MgC₂ at high pressure. All authors contributed to writing and reviewing the manuscript.

Competing interests
The authors declare no competing interests.

Additional information
Correspondence and requests for materials should be addressed to D.Y.K.

Reprints and permissions information
is available at www.nature.com/reprints.

Publisher’s note
Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access
This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article’s Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article’s Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2019