Mechanical and Volumetric Fracturing Behaviour of Three-Dimensional Printing Rock-like Samples Under Dynamic Loading

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Abstract
Heterogeneous rock contains numerous pre-existing three-dimensional (3D) cracks, which control its mechanical and fracturing properties. Considerable effort has been devoted to studying the volumetric fracturing behaviour of rock under static loading conditions. Although rock masses are often subject to dynamic impacts such as earthquakes and blasting, the mechanical and volumetric fracturing behaviour of rock under dynamic loading is still poorly understood. In this paper, dynamic laboratory tests were performed on 3D-printed artificial rock samples with 3D embedded flaws created during three-dimensional printing (3DP), with the aim of studying the volumetric fracturing and mechanical properties of these samples under impact with high strain rate. The results show that the dynamic compressive strength and the tangent modulus decrease with an increasing number of flaws, but have very limited effects on the ratio of the fracture initiation stress of the first crack to the peak stress of the sample, the maximum axial strain of the sample and the volumetric fracturing behaviour of the sample. The tensile failure of a sample is caused by the continuous extension of wing cracks from the outer flaw tips. The mechanical and volumetric fracturing behaviour of samples with 3D embedded flaws are strain rate dependent. The tangential modulus and the ratio of the fracture initiation stress of the crack to the peak stress increase significantly when the loading type changes from static compression to dynamic compression. Under dynamic compression, wing cracks can continuously extend to the sample ends, whereas under static compression, wing cracks can intermittently extend only a limited distance. Moreover, the fracturing behaviour of 3D flaw differs from that of 2D flaws under dynamic loading. Under high strain rate loading, wing cracks generated at 3D flaw tips lead to splitting failure of the sample, while shear cracks formed at 2D flaw tips result predominant shear failure of the sample. The findings in this paper could facilitate a better understanding of rock failure subjected to dynamic loading conditions.

Keywords Volumetric fracturing · 3D embedded flaw · 3D printing · Wing crack · Dynamic loading

1 Introduction

Rock is heterogeneous and contains a multitude of pre-existing micro- and macro-defects, e.g., voids, inclusions and joints. The collapse of rock materials subjected to external or internal loads is due to both the activation of these pre-existing defects and the generation of new micro- and macro-cracks (Peng and Johnson 1972). Thus, it is of great academic interest and engineering significance to understand volumetric fracturing behaviour in rock and rock masses. Volumetric fracturing refers to the three-dimensional (3D) fracturing process characterized by large-scale, multi-fracture, high-intensity damage and energy release, resulting from the interaction, growth, and division of the internal defects of the rock (Xie et al. 2016; Zhou et al. 2019).
Rock heterogeneity makes it difficult to predict and analyse crack growth in rocks using theoretical methods (Liu et al. 2004). In contrast, as a fundamental and effective method, laboratory experiments enable scholars to directly monitor and study crack growth in rocks under various conditions. To date, a large number of experimental investigations have been performed to study two-dimensional (2D) crack initiation and coalescence in rocks subject to static loading (Bobet and Einstein 1998; Feng et al. 2009; Wong et al. 2001; Yang et al. 2009; Zhang et al. 2018). Wong and Einstein (2009) analysed crack initiation, propagation and coalescence in gypsum samples and marble samples each containing a single 2D flaw (henceforth defined as pre-existing defect) subject to static compression. They noted that the cracks that initiate from the pre-existing 2D flaws can be characterized into three main types, i.e., tensile cracks, shear cracks and mixed tensile-shear cracks. Gale et al. (2007) investigated the importance of natural pre-existing defects for hydraulic fracture treatment in enhanced geothermal systems and noted that even the smaller sealed defects serve as weak planes that significantly affect the propagation of the hydraulic fractures.

Rock is often subjected to dynamic loads such as earthquakes and blasting (Deng et al. 2015; Yim et al. 1980). 2D crack growth in rock and gypsum samples has been investigated under not only static loading conditions but also dynamic loading conditions in recent years (Li et al. 2016, 2017; Li and Wong 2012; Zou and Wong 2014). Zou and Wong (2014) extended the research of cracking processes in rock samples with 2D flaws from static loading conditions to dynamic loading conditions. Li et al. (2017) performed experiments on plate samples with 2D flaws using a split Hopkinson pressure bar (SHPB) device. In these studies, shear cracks always occurred prior to tensile cracks in the SHPB tests, and the tensile cracks were often suppressed by the dominant propagation of shear cracks. Additionally, it was reported that the presence of 2D flaw may change the failure mode of marble from splitting-dominated to shear-dominated failure under dynamic loading (Li et al. 2017; Zou and Wong 2014).

Naturally, flaws in rock are 3D features. To date, many studies have been conducted to study the growth of 3D surface flaws (Adams and Sines 1978; Liang et al. 2012; Wong et al. 2004; Zhou et al. 2018a). A 3D surface flaw is a flaw that is partially embedded in rock (Yin et al. 2014). Cannon et al. (1990) performed static uniaxial compression experiments on polymethyl methacrylate (PMMA) samples containing a half-penny shaped 3D surface flaw. They reported that crack propagation from pre-existing 3D surface flaws differs from that in 2D flaws. Wing cracks were observed to form sail-like geometries, as they wrapped around the edges of 3D surface flaws. Yin et al. (2014) carried out an experimental study on granite samples fabricated with two 3D surface flaws under static compression with the aim of investigating the coalescence mechanism between the pre-existing flaws. Two major coalescence mechanisms in the fracturing process of the 3D surface flaws were identified: the linkage between the surface flaws and the coalescence between the petal cracks within the rock.

In addition to 3D surface flaws, numerous 3D embedded flaws also exist in rock. However, studies on the volumetric fracturing behaviour of 3D embedded flaws are quite limited. One of the reasons for this limitation is the difficulty in producing samples containing 3D embedded flaws. By embedding a thin aluminium foil plate inside the model when preparing rock-like samples, Dyskin et al. (2003) prepared resin samples with 3D embedded flaws. They investigated volumetric fracturing using 3D-printed artificial rock samples containing 3D embedded flaws under static loading conditions. By comparing the wing crack growth and failure pattern of 3DP resin samples each containing a single flaw with the inclination angle of 30° with previous findings presented by Germanovich et al. (1994) and Dyskin et al. (2003) under similar loading conditions, Zhou et al. (2019) validated the application of flawed resin samples produced via 3DP to the investigation of crack propagation and coalescence under compression.

However, no study on volumetric fracturing of 3D embedded flaws under dynamic loading conditions has been reported so far. Therefore, it is necessary to study the volumetric fracturing and mechanical behaviour of rock or rock-like samples with 3D embedded flaws subjected to dynamic loading.

In this paper, dynamic tests were performed on artificial rock samples with 3D embedded flaws prepared using 3DP, with the aim of investigating volumetric fracturing and the mechanical behaviour of rock under dynamic compression. The influences of the loading type and number of flaws on the mechanical properties and volumetric fracturing behaviour were analysed. A comparison between the volumetric fracturing and mechanical behaviour of 3D embedded flaws under static and dynamic loading conditions and a comparison between the 2D and volumetric fracturing behaviour of 3D embedded flaws subjected to dynamic loading were performed. This study is a further study of the volumetric fracturing and mechanical properties of resin-based 3D-printed samples containing 3D embedded flaws under static compression (Zhou et al. 2019). The findings in this paper could facilitate a better understanding of the failure behaviour and mode of rock under dynamic loading conditions. Notably,
in this study, the terms “flaw” and “crack” refer to the pre-existing defects and the newly generated cracks (e.g., wing and secondary cracks) inside the sample, respectively, and “fracture” refers only to the crack evolution process.

2 Experimental Setup

2.1 Sample Preparation

Details of the sample preparation and 3DP are provided in our previous studies (Zhou et al. 2019; Zhu et al. 2018). Figure 1 presents the sample geometry, where the flaw angles ($\alpha$) of the samples with one and two flaws are 30° and 45°, respectively; for samples with two flaws, the bridge angle ($\beta$) between the inner tips of the two flaws is 105°. Table 1 provides specific information on the 3D-printed samples, where samples S-1 and D-1 represent static and dynamic uniaxial compression tests conducted on 3D-printed samples containing a single flaw, respectively. The aperture of the flaws is 2 mm. Notably, because one of the advantages of 3DP is its high repeatability in sample preparation and thus experimental results (Gell et al. 2019; Jiang and Zhao 2015; Jiang et al. 2016), only one sample was tested per geometry.

2.2 Apparatus

The dynamic tests were performed with a modified SHPB device (Li et al. 2005; Liao et al. 2016). During testing, the cone-shaped striker impacted the input bar, generating a half-sine incident pulse (time duration of approximately 372 ms).
250 μs) that propagated along the steel bars. When the incident wave $\varepsilon_I(t)$ arrived at the bar-sample interface, it split into a reflected wave $\varepsilon_R(t)$ and a transmitted wave $\varepsilon_T(t)$. When the stresses at the sample sides were approximately equal, a stress balance was achieved within the sample. Then, according to the ISRM-suggested method (Zhou et al. 2012), the dynamic compressive stress $\sigma(t)$ and strain $\varepsilon(t)$ can be determined as

$$\sigma(t) = \frac{A_e E_e}{2A_s} [\varepsilon_I(t) + \varepsilon_R(t) - \varepsilon_T(t)]$$

$$\varepsilon(t) = \frac{C_e}{L_s} \int_0^t [\varepsilon_I(t) - \varepsilon_R(t) - \varepsilon_T(t)] dt$$

where $A_e$, $C_e$, and $E_e$ are the cross-sectional area (1963.5 mm²), longitudinal wave velocity (5400 m/s), and elastic modulus (240 GPa) of the steel bars, respectively. $A_s$ and $L_s$ refer to the cross-sectional area (mm²) and length (mm) of the sample, respectively.

In the SHPB tests, two FASTCAM SA1.1 high-speed cameras (HSCs) were utilized to monitor real-time volumetric fracturing in two orthogonal directions. The HSCs were automatically triggered by the input signal, and the frame rate and resolutions were 125,000 frames per second (fps) and 256 × 128 pixels, respectively. Accordingly, the time interval between the two adjacent high-speed images was 8 μs. Notably, as the HSC’s storage space was 8 GB, the recording time of the HSC was restricted to 0.56 s during the SHPB tests.

### 3 Results

#### 3.1 Mechanical Properties

Figure 2 presents the dynamic stress–strain curves of the samples with one and two 3D embedded flaws. The results show that when the number of flaws increased from one to two, the dynamic compressive strength decreased from 113.7 to 95.9 MPa, a reduction of 15.7%; the tangent modulus at half the peak stress decreased by 11.8%; and the maximum axial strain at failure of the sample slightly reduced by 4.8%. However, the shapes of the dynamic stress–strain curves of the single-flawed and double-flawed samples are similar, which indicates that the number of flaws may not affect the dynamic stress–strain behaviour of the samples.

#### 3.2 Volumetric Fracturing in 3D-printed Samples

The images of the volumetric fracturing of the flawed samples are shown in Fig. 3. Regarding the sample containing a single flaw, when the dynamic stress reached 91% of the peak stress (113.7 MPa), at a loading time of approximately 104 μs, symmetrical wing cracks almost simultaneously initiated at the flaw tips (Fig. 3a). With increasing load, the wing cracks slightly wrapped around the flaw boundary and synchronously propagated at a relatively steady rate towards the sample ends, eventually splitting the 3D-printed resin sample into strips, as shown in Fig. 3c.

For the double-flawed sample, two wing cracks almost simultaneously initiated at the inner tip of the upper flaw and outer tip of the lower flaw, respectively, when the dynamic stress reached approximately 90% of the peak stress (95.9 MPa) (see Fig. 3b). The extension and interaction of the wing cracks from inner tips of the flaws resulted in the coalescence of flaws (Fig. 3b). Subsequently, a wing crack initiated at the outer tip of the upper flaw when the loading time reached approximately 104 μs. Then, the wing cracks at the outer tips of the upper and lower flaws continuously extended along the dynamic impact direction towards the ends of the sample as the loading time increased to approximately 200 μs. However, propagation of the coalescent wing cracks at the inner tips of the flaws was constrained by the dominant evolution of the wing cracks at the outer tips of the flaws and could not expand farther (Fig. 3b). The continuous extension of wing cracks at the outer tips of the flaws finally resulted in the tensile failure of the sample (Fig. 3d). Figure 3 also shows that the fracturing and failure behaviour between the single-flawed and double-flawed samples are similar, except that crack coalescence occurs in the double-flawed sample. This similarity indicates that the number of flaws exerts an insignificant influence on the fracture and failure mode of the samples containing 3D embedded flaws.
3.3 Crack Propagation Velocity

With the aid of an HSC, real-time fracturing can be monitored during testing; therefore, the crack propagation velocity ($V_c$) can be determined. Notably, $V_c$ is an average velocity value within a time interval (i.e., 8 μs) and is defined as the ratio of the propagation length over the time interval between two high-speed images. Figure 4 illustrates crack propagation velocities in samples with single and double 3D embedded flaws subjected to dynamic compression. In general, the $V_c$ of wing cracks tends towards a stable value, and the terminal speed is slightly lower than the initial speed. For example, the initial and terminal propagation speeds of the wing crack at the outer tip of the upper flaw in the double-flawed samples were approximately 350 and 319 m/s, respectively. However, the maximum $V_c$ of the single-flawed sample (650 m/s) was nearly 30% higher than that of the double-flawed sample (451 m/s).

4 Discussion

4.1 Comparison of the Mechanical Behaviour of 3D Flawed Samples Under Dynamic and Static Loading Conditions

Table 1 lists the mechanical properties of the tested 3D-printed samples under static compression (Zhou et al. 2019) and dynamic compression. The mechanical properties are dependent on the loading conditions. For instance, from static compression to dynamic compression, the axial strain at failure decreased by 40.1% and 24.4% for samples containing one and two flaws, while the tangent modulus increased by 52.4% and 38.4%, respectively. The strain rate effect on the mechanical properties of 3D-printed resin is believed to be associated with secondary molecular processes, which means that increasing the loading rate decreases the molecular mobility of polymer chains, thus stiffening the chains (Richeton et al. 2006). This mechanism is similar to the temperature effect on the compressive strength of polymers, where a very low environmental temperature strongly constrains the movement of polymer chains, yielding a higher compressive strength of polymers at temperatures below the glass transition temperature (Zhou and Zhu 2018).

In the SHPB tests, the impact of the number of flaws on the initiation stress ($\sigma_1$) of the first crack to form in 3D-printed resin samples is consistent with that on the peak stress ($\sigma_C$), i.e., $\sigma_1$ decreases with an increasing number of flaws. However, the effect of the number of flaws on the ratio of $\sigma_1/\sigma_C$, which was approximately 0.9 for the flawed samples in the dynamic tests, is negligible. This is similar to the findings in static compression tests, where $\sigma_1/\sigma_C$ also seemed to be unaffected by the number of flaws, although $\sigma_1$ in samples containing two 3D embedded flaws is lower than that in samples with a single flaw (Zhou et al. 2019). This lack of variability is possibly because $\sigma_1/\sigma_C$ is an intrinsic property of a rock material, which may be unaffected by the sample geometry and boundary conditions (Bieniawski 1967; Diederichs et al. 2004; Xu et al. 2018). In addition, $\sigma_1$ in the dynamic tests increased by approximately 73% and 80% for samples with single and double flaws, respectively, compared with the results of the static tests, which means that a strong strain rate effect on $\sigma_1$ exists. This effect is consistent with the results found in the numerical study of the effect of loading rate on crack initiation stress, despite the difference in the magnitude of increase (Zhang and Wong 2013). The reason why $\sigma_1$ is higher under dynamic loading than under static loading is mainly because the crack initiation toughness is rate dependent and increases with the strain/loading rate (Jajam et al. 2013; Zhou et al. 2018b).

4.2 Comparison of the Fracturing Behaviour of 3D Flawed Samples Under Dynamic and Static Loading Conditions

Wing crack growth in samples with 3D embedded flaws is significantly different under dynamic and static loading conditions, as shown in Fig. 5. Under dynamic compression, wing cracks rapidly propagated and continuously extended to the ends of the sample, finally resulting in a splitting failure of the sample, whereas under static compression, wing cracks intermittently formed but could extend only a limited distance, and the final failure was caused by secondary cracks (the secondary cracks initiated later than the wing cracks). This difference possibly arises because under dynamic loading, after crack initiation, the stress intensity factor (SIF) remains approximately constant for a certain time, which drives wing cracks to grow continuously until sample failure (Li et al. 2000), whereas in the static compression test, the wrapping of wing cracks around the flaw edges restricts the wing cracks from propagating a long distance (Dyskin et al. 2003).

In addition, unlike under static compression (Fu et al. 2016; Zhou et al. 2019), under dynamic loading, anti-wing cracks (see Fig. 5b) did not initiate in the flawed samples. This phenomenon may be attributed to stress wave propagation altering the stress field around the 3D flaw tips where anti-wing cracks could initiate under static compression. This mechanism is similar to the mechanism controlling the initiation of shear cracks before tensile cracks from the 2D flaw tips under dynamic loading (Li et al. 2017; Li and Wong 2012). Under dynamic loading, the arrival of the stress wave generates compressive stress where tensile cracks often initiate under static loading, temporarily constraining the development of tensile cracks (Li and Wong 2012).
Moreover, the wing crack propagation speed approached a relatively stable value under dynamic loading conditions, whereas under static compression, $V_c$ of the secondary cracks quickly decreased with increasing propagation time, as shown in Fig. 6. The difference in the decrease in velocity between the dynamic and static tests is probably because the dynamic stress could be roughly maintained at the peak stress level during the entire crack propagation process, even within a certain period during the unloading process in the SHPB tests, whereas under static loading conditions, due to the rapid unloading of the axial stress caused by the burst-like failure of the sample, the peak stress cannot be maintained after initiation of the secondary cracks (Gao et al. 2015; Li et al. 2017). However, the maximum $V_c$ (650 m/s) in the SHPB tests was lower than that in the static compression tests (977 m/s). The difference may be attributed to the discrepancy between the SIFs of samples under dynamic and static compression conditions. Take the sample containing a single 3D embedded flaw as an example. In the SHPB and static compression tests, the stresses applied to the sample were 102.8 MPa (approximately 90% of the peak stress) and 107.3 MPa, respectively, when the wing cracks and secondary cracks initiated at the flaw tips or around the flaw edges. As the SIF is related to the stress around/ahead of the flaw/crack tips (Benz and Sander 2014; Rossmanith 1983), it is therefore inferred that the higher loading stress under static compression may result in greater SIFs around the flaw/crack tips than those resulting from the SHPB tests. Moreover, according to Charles’ law, $V_c = a b^n$, in which $a$ is constant, $b$ is the SIF, and $n$ is the stress corrosion index whose range is 20–60 for geomaterials (Atkinson and Meredith 1987a, b; Main et al. 1989). Consequently, the greater SIFs in as specimen under static loading may result in more rapid crack propagation.

![Fig. 3 Volumetric fracturing of the flawed samples: a and b Progressive fracturing of the sample with single and double 3D embedded flaws, respectively; c and d failure fragments after dynamic compression tests. FS and SS represent photographing from front and side surfaces of the resin sample, respectively. The time marked in (a) and (b) is the absolute time from the start of dynamic loading. WC indicates wing cracks. Red arrows indicate the newly generated wing cracks (color figure online)](image)

![Fig. 4 Crack propagation velocities in dynamic compression tests](image)

![Fig. 5 Wing crack growth in: a Dynamic test; and b static test. The crack growth in static test was modified after Zhou et al. (2019). The time marked in (a) is the absolute time from the start of dynamic loading. The stress marked in (b) is the initiation stress of the wing and anti-wing cracks during static compression](image)

![Fig. 6 Crack propagation velocities in dynamic and static compression tests. The crack propagation velocity data in static test is modified after Zhou et al. (2019). SC refers to secondary crack](image)
4.3 Comparison of the Fracturing Behaviour of 2D and 3D Embedded Flaws Under Dynamic Loading Conditions

Under dynamic compression, the volumetric fracturing behaviour among samples with single 2D and 3D embedded flaws is distinct. For a sample with a 2D flaw, the dominant shear crack results in the final failure with an X-shaped pattern, as shown in Fig. 7a (Zou and Wong 2014; Li et al. 2017). Li and Wong (2012) reported that high compressive stresses at 2D flaw tips lead to the rapid initiation and propagation of shear cracks. In contrast, wing cracks initiate at 3D embedded flaw tips and continuously extend towards the sample ends (Fig. 7b). This failure mode may be due to the sliding of flaw surfaces, which resembles that observed during 2D crack propagation under static compression (Nemat-Nasser and Horii 1982).

5 Conclusions

Dynamic impact tests were conducted on 3D-printed resin samples with 3D embedded flaws using a SHPB device. The mechanical and fracturing behaviour of the flawed-samples were investigated. The difference between the mechanical and volumetric fracturing behaviour of a 3D embedded flaw under static and dynamic loading conditions and between volumetric fracturing and 2D fracturing behaviour under dynamic loading conditions were determined. The main conclusions of this paper are summarized as follows.

1. The loading type influences the mechanical properties of samples containing 3D embedded flaws. From static compression to dynamic compression, $\sigma_C$ increases by approximately 8%, and the maximum axial strain decreases by 40.1% and 24.4% for samples with one and two 3D embedded flaws, respectively. The Young’s modulus, $\sigma_\tau$, and $\sigma_1/\sigma_C$ significantly increase when the loading type changes from static to dynamic.

2. The number of flaws has a very limited effect on the ratio of $\sigma_1/\sigma_C$ and the volumetric fracturing behaviour of the flawed samples. The ratio of $\sigma_1/\sigma_C$ is approximately 90% for the flawed samples under dynamic loading. The propagation behaviour of wing cracks, the failure mode and fragment geometry after failure are almost identical between the samples with one and two 3D embedded flaws.

3. The volumetric fracturing of 3D embedded flaws is strain rate dependent. Under dynamic compression, wing cracks can continuously extend toward the sample ends, resulting in splitting failure of the sample, whereas wing cracks could only intermittently grow a limited distance under static compression. The crack propagation speed approaches a stable value under dynamic loading, but the peak crack propagation velocity in the SHPB tests is lower than that in the static compression tests.

4. Under dynamic loading, the fracturing of samples with 3D embedded flaws is distinct from that of the samples with 2D flaws. Wing cracks initiation at the 3D flaw tips lead to splitting failure of the sample, while the dominant shear cracks initiating at 2D flaw tips result in shear failure of the sample.

Although the preliminary findings of this study could facilitate a better understanding of rock failure behaviour, the resin-based artificial rock samples cannot completely reflect the mechanical and fracturing behaviour of hard and brittle rocks due to the limitations (e.g., low Young’s modulus and almost no porosity) of the 3D-printed resin material. Therefore, further improvements, e.g., enhancing the 3D-printed resin’s elastic modulus and brittleness or implanting the internal micro- and macro-cracks geometries of natural rock into 3D-printed resin-based samples, are needed so that the properties of the 3D-printed resin approach those of actual rocks. In addition, further efforts are needed to more precisely and quantitatively characterize the 3D stress field around pre-existing flaws and the progressive fracture process subjected to different loading conditions by means of, for example, the digital image correlation and photoelastic methods.

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### Appendix

#### Supplementary videos

Four supplementary videos taken by the HSCs are supplied for better understanding 3D internal crack growth under dynamic compression. Notably, all of the videos are played at three frames per second and can be viewed through the Nature Springer suggested website: https://doi.org/10.6084/m9.figshare.7525001.v1.

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