Structural complexities and sodium-ion diffusion in the intercalates Na$_x$TiS$_2$: move it, change it, re-diffact it†

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After momentary attention as potential battery materials during the 1980s, sodium titanium disulphides, like the whole Na–Ti–S system, have only been investigated in a slapdash fashion. While they pop up in current reviews on the very subject time and again, little is known about their actual crystal-structural features and sodium-ion diffusion within them. Herein, we present a short summary of literature on the Na–Ti–S system, a new synthesis route to Na$_{x}$TiS$_2$-3R, and results of high-temperature X-ray and neutron diffractometry on this polytype, which is stable for medium sodium content. Based thereon, we propose a revision of the crystal structure reported in earlier literature (missed inversion symmetry). Analyses of framework topology, probability-density functions, and maps of the scattering-length density reconstructed using maximum-entropy methods (all derived from neutron diffraction) reveal a honeycomb-like conduction pattern with linear pathways between adjacent sodium positions; one-particle potentials indicate associated activation barriers of ca. 0.1 eV or less. These findings are complemented by elemental analyses and comments on the high-temperature polytype Na$_{0.9}$TiS$_2$-2H. Our study helps to get a grip on structural complexity in the intercalates Na$_x$TiS$_2$: caused by the interplay of layer stacking and Na–Ti–vacancy ordering, and provides first experimental results on pathways and barriers of sodium-ion migration.

**Introduction**

Layered lithium-ion conductors are a successful substance class for energy storage, e.g., in modern consumer electronics. Much effort has been put into research on their sodium congeners as possible high-voltage, low-cost alternatives—so far, to no commercial avail. Although alkali transition-metal dichalcogenides are not a recent focus, they pop up in current reviews time and again. 

As first Na–Ti–S compound, uncharacterised “Na$_2$TiS$_3$” was reported in a process patent to have an X-ray pattern similar to the also uncharacterised “Li$_2$TiS$_3$” and “K$_2$TiS$_3$” in 1960. Five years later, Rüdorff described the first layered structure in the system and wrongly assigned it the NaHF$_2$ type [modern prototype according to the Inorganic Crystal Structure Database (ICSD): CrOOH[R$_3$m]]—not yet aware of its hydrogen positions. Reliable sodium positions were not established before the first systematic study in 1971. Today, five kinds of Na–Ti–S compounds can be distinguished:

- The structure of the intercalates Na$_x$TiS$_2$ with 0 < $x$ ≤ 1 consists of $\frac{2}{3}$[TiS$_2$] sandwich layers with sodium ions in between. These compounds exist in four different structurally elucidated stacking variants: the polytypes 2H, 3R$_3$, 3R$_2$, and 6H (see Table 1 and Fig. 1). With these materials, the remaining article is concerned.

- The excess-titanium compounds Na$_x$Ti$_{1+\delta}$S$_2$ with 0 < $\delta$ ≤ 1 and 0 < $x$ ≤ 1 – 3$\delta$ contain additional titanium ions in intercalation layers, which hinder diffusion of the otherwise mobile sodium ions.

- Contrary to its first assignment, Na$_3$TiS$_4$ ≡ Na$_2$.5Ti$_{0.5}$S$_2$ exhibits a structure homeotypic to Na$_2$SnS$_3$, not K$_2$TiS$_3$. At closer inspection, it crystallises in a sixfold superstructure of the 3R polytypes with an average structure akin to Na$_3$TiS$_2$-3R$_2$. It represents an OD structure with alternating pure sodium and mixed sodium-titanium layers, constituting a distorted NaCl homeotype. Upon heating, Na$_3$TiS$_4$ decomposes to NaTiS$_2$-2H under oxidation of sulphide. The compound dubbed “3R(II)-NaTiS$_2$” most probably is a sodium-deficient variant of Na$_2$TiS$_3$: 

- The electrochemically intercalated compounds Na$_x$TiS$_2$ contain sulphide as well as disulphide ions and are formed from TiS$_2$ via disulphide (instead of titanium) reduction and
concomitant bond cleavage.\textsuperscript{17} They have neither been studied ex situ nor structurally characterised.

- Structurally complex Na\textsubscript{x}TiS\textsubscript{2} has only been reported once in a book of poster abstracts. It is said to crystallise in the orthorhombic space group \textit{Fdd2} with \( a = 38.49(2), b = 59.36(3), c = 7.033(2) \) \( \text{Å} \), and \( Z = 120 \). The unit cell allegedly contains eight crystallographically independent TiS\textsubscript{4} tetrahedra, which are arranged in triple slabs parallel to (010).\textsuperscript{18}

The polytype, which Na\textsubscript{x}TiS\textsubscript{2} assumes, depends on \( x \) as well as on temperature. In the past, inconsistencies in reported stability ranges were discussed and \textit{bona fide} best values were derived acknowledging synthetic (chemical vs. electrochemical intercalation) and analytical differences (diffraction patterns vs. discharge curves).\textsuperscript{11} The only reported thermally induced phase transformations are the entropy-driven order–disorder transition for small \( x \) according to eqn (1) and the irreversible reconstitution of the (at ambient temperature metastable) 2\( H \) phase for large \( x \) following eqn (2).\textsuperscript{8} Transformations to the 2\( H \) form from either of the 3\( R \) polytypes were not observed and an irreversible transition between the latter was attributed to decomposition via sodium loss.

\[
\text{Na}_{0.7}\text{TiS}_2 - 6R \xrightarrow{430 \degree C} \text{Na}_{0.3}\text{TiS}_2 - 3R_1 \tag{1}
\]

\[
\text{NaTiS}_2 - 2H \xrightarrow{\text{slow cooling}} \text{NaTiS}_2 - 3R_2 \tag{2}
\]

Ordering of sodium ions and vacancies was detected in electrochemically intercalated TiS\textsubscript{2} single crystals, where \( 2 \times 2, 2 \times \sqrt{3}, \text{and} \sqrt{3} \times \sqrt{3} \) superstructures may occur. For \( x < 0.11 \), a stage-3 compound with a \( \sqrt{3} \times \sqrt{3} \) superstructure of a 3\( R \) phase containing sodium in trigonal-antiprismic coordination was discovered but not conclusively characterised.\textsuperscript{19,20} The existence of such superstructures was later rationalised,\textsuperscript{21} comprehensively computed and explained.\textsuperscript{22}

Data on sodium-ion diffusion in Na\textsubscript{x}TiS\textsubscript{2} are sparse. The chemical diffusivity at ambient temperature is specified as \( \approx 10^{-9} \text{ cm}^2 \text{ s}^{-1} \) for \( 0.25 < x < 0.6 \) (probably polytype 3\( R_1 \)).\textsuperscript{23,24} The much higher value of \( 10^{-7} \) to \( 10^{-6} \text{ cm}^2 \text{ s}^{-1} \) for \( x < 0.45 \), which was only reported once, remains questionable.\textsuperscript{25} Unfortunately, migration barriers were merely computed for Na\textsubscript{0.7}TiS\textsubscript{2}-3\( R_1 \) (0.19 eV)\textsuperscript{26} and a hypothetical Na\textsubscript{0.7}TiS\textsubscript{2} with O1 coordination/layer sequence (1.02 eV).\textsuperscript{27} Trigonal-antiprismic coordination, as realised in the polytype 3\( R_2 \), seems to hinder diffusion, so that diffusivities and migration barriers should be lower and higher in it, respectively, than in the other polytypes.\textsuperscript{28}

Our dealing with these materials is less inspired by prospective application as an ion conductor than by improving the grasp of ion diffusion in them and a comparing them to their lithium congeners. This is why, herein, we report on the revised crystal structure of sodium and sodium-ion diffusion in Na\textsubscript{3}TiS\textsubscript{2}-3\( R_1 \) based on high-temperature neutron diffraction (ND). The study comprises

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\begin{table}[h]
\centering
\caption{Crystal-structurally characterised polytypes of Na\textsubscript{0.7}TiS\textsubscript{2} (0 < x < 1) from literature}
\begin{tabular}{|c|c|c|c|c|}
\hline
Polytype & 2\( H \) & 3\( R_1 \) & 3\( R_2 \) & 6\( R \) \\
\hline
Stability range\textsuperscript{a} & High-temperature phase & 0.40(2) < \( x \) < 0.64(2) & 0.75(2) < \( x \) \leq 1 & 0.15(2) < \( x \) < 0.25(2) \\
& & (0.35 < \( x \) < 0.72) & (0.70 < \( x \) \leq 1) & (0.10 < \( x \) < 0.33) \\
Best characterised & \( x = 1 \) (ref. 7) & \( x = 0.55 \) (ref. 6) & \( x = 1 \) (ref. 6) & \( x = 0.3 \) (ref. 6 and 8) \\
representative & & & & \\
Coordination polyhedron\textsuperscript{b} & Na\textsuperscript{6p} & Na\textsuperscript{6p} & Na\textsuperscript{6ap} & Na\textsuperscript{6p} \\
& Ti\textsuperscript{60} & Ti\textsuperscript{60} & Ti\textsuperscript{60} & Ti\textsuperscript{60} \\
Stage & 1 & 1 & 1 & 2 \\
Space-group type & \( \text{P}_6_3/mmc \) & \( \text{R}^3\text{m} \) & \( \text{R}^3\text{m} \) & \( \text{R}^3\text{m} \) \\
Structure prototype (ICSD) & Na\textsubscript{3}TiS\textsubscript{2} & CuCrSe\textsubscript{2}\textsuperscript{d} & Delafossite-NaCrS\textsubscript{2} & Ag\textsubscript{1.5}Nb\textsubscript{6}S\textsubscript{12} \\
Delmas’ notation\textsuperscript{10} & P\textsubscript{2} & P\textsubscript{3} & O\textsubscript{3} & O\textsubscript{1}–P\textsubscript{3}\textsuperscript{c} \\
Anion stacking & AABB & AABBC & ABCABC & ABABBCBCCACA \\
Sodium-ion order\textsuperscript{b} & Disordered over all voids & Ordered in half of the voids & Ordered in half of the voids & Disordered over all voids \\
\textsuperscript{a} Best values;\textsuperscript{11} reported maximum extents in parentheses. \textsuperscript{b} Within occupied sodium layers. \textsuperscript{c} Corrected for a missed mirror plane,\textsuperscript{6} originally given as R3. \textsuperscript{d} Corrected in May 2019 (see Table S1),\textsuperscript{13} formerly ambiguously given as CuCrSe\textsubscript{2} usivity at ambient temperature is speci
\hline
\end{tabular}
\end{table}

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\begin{figure}[h]
\centering
\includegraphics[width=\textwidth]{fig1.png}
\caption{Crystal structures of the Na\textsubscript{0.7}TiS\textsubscript{2} polytypes; left to right: 2\( H \), 3\( R_1 \), 3\( R_2 \), and 6\( H \). Atoms are plotted with arbitrary radii, unit cells are depicted in black. Note: depending on \( x \), sodium positions may not be fully occupied.}
\end{figure}
topological analyses, the visualisation of diffusion pathways using the probability-density function (PDF) and maps of the scattering-length density (SLD) reconstructed via maximum-entropy methods (MEM), as well as an evaluation of migration barriers using the effective one-particle potential (OPP). Results are complemented by temperature-dependent X-ray diffraction (XRD), elemental analyses, and comments on the related compound Na_{0.8}TiS_2·2H_2O.

**Experimental**

**Analytical method**

Hydrogen and sulphur contents were determined using a “Thermo Finnigan Flash EA 1112” analyser. Sodium was determined via optical emission spectroscopy with an inductively coupled plasma (ICP-OES) using a “Thermo Fisher Scientific iCAP 6300 Duo” evaluating the emission line at 589.5 nm.

The numbers of titanium ions per formula unit (with respect to sodium) were determined via wavelength-dispersive X-ray fluorescence (XRF) measurements on a “PANalytical Axios” spectrometer evaluating the intensities for the Na-Kα and Ti-Kα transitions. Samples were prepared from ca. 100 mg of analyte and 100 mg wax (Hoechst wax) performing uniaxial pressing (PerkinElmer hydraulic hand-press). Intensity ratios and sodium intensities were calibrated against linear regressions for three different mixtures of Na_2S and TiS_2 with R(Na/Ti) = N(Na) = 0.4, 0.7, and 1.0 to account for instability/correlation of the titanium signal.

**Synthetic procedures**

**Synthesis of TiS_2·17.** Polycrystalline TiS_2·17 was prepared via direct synthesis from titanium (ca. 3.5 g, 1.00 eq.; Merck) and sulphur (2.10–2.20 eq.; Merck) in an evacuated silica ampoule. In a chamber furnace, the ampoule was heated to 400 °C with 25 °C h^{-1} and held there for 40 h to achieve complete reaction. The temperature was then elevated to 600 °C and kept for 99 h, after which the oven was turned off to cool to room temperature. Residual sulphur was evaporated in vacuo at 160 °C. Identity and purity of the resulting greyish black powder with metallic lustre (typically ca. 8 g) were confirmed using XRD.

**General synthesis of Na_{0.9}TiS_2.** TiS_2·17 (ca. 3.5 g, 1.00 eq.) and Na_2S (0.25 or 0.45 eq.; Sigma-Aldrich, ≥97%) were ground together in an agate mortar. The mixture was placed in a corundum crucible, which was subsequently heated to 700 °C with a rate of 400 °C h^{-1} in a tube furnace under H_2S atmosphere (99.5%, Air Liquide), kept at that temperature for 2 h, and then cooled to ambient temperature with a rate of 200 °C h^{-1}. The product was removed in a dinitrogen counter-flow and stored under argon. The reaction afforded Na_{0.9}TiS_2 (yield: ca. 99%) as greyish black, microcrystalline powder with a metallic lustre for high sodium content.

Found: H, 1.0(1); Na, 8.8(1); S, 48(1); N(Ti) = 1.0(2). Calc. for Na_{0.8}TiS_2·2H_2O: H, 0.8; Na, 8.7; S, 48.4%; N(Ti) = 1.0.

**X-ray diffraction**

Measurements for phase identification and Rietveld refinement were carried out at ambient temperature on a “PANalytical X’Pert PRO MPD” diffractometer equipped with a “PIXcel” detector using nickel-filtered Cu-Kα radiation in Bragg–Brentano (θ–θ) geometry (see Fig. S1| for details). Temperature-dependent diffractograms were recorded in dinitrogen atmosphere on a “Rigaku SmartLab 3 kW” system using nickel-filtered Cu-Kα radiation.

**Neutron diffraction**

Measurement was carried out at Institut Laue-Langevin (ILL) using the high-resolution two-axis diffractometer D2B with Ge(335)-monochromated constant-wavelength radiation (λ = 1.594 Å) in Debye–Scherrer geometry.⁹,¹⁰ Compacted powder samples were mounted in a vacuum high-temperature furnace inside a vanadium can (d = 13.9 mm, h = 49.8 mm). Measurements were carried out at ambient temperature, 300, 600, and 700 °C. Data were recorded with an array of 128 ³He tubes (height: 300 mm), yielding a final range of 0.10 ≤ 2θ ≤ 159.90° with Δ(2θ) = 0.05°. Initial Le-Bail fits and following Rietveld refinements were carried out using JANA2006.¹¹ Neutron data were analytically corrected for absorption (cylindrical sample) and stripped of the irregular onset below 6.5° and, if necessary, cut-off reflections above 155°. Peak profiles were fitted with a pseudo-Voigt function using the Thompson–Cox–Hastings approach (Gaussian parameters U, V, and W; Lorentzian parameter Χ). A zero-shift correction and an asymmetry correction according to Howard were applied.¹² The background was modelled using ten Legendre polynomials interpolating between manually defined points. Relatively weak reflections of a by-phase assuming the space-group type Fm3m (probably steel from the sample environment) were treated with an appropriate Le-Bail fit (separate profile parameters).

As a starting point for Rietveld refinement, an atomic model of Na_{0.55}TiS_2 was imported from the ICSD and adjusted to reflect the actual cell content and symmetry.¹³ Anisotropic displacement parameters were refined for all atoms except for Ti1 at 18 °C, the very small displacement of which had to be constrained to isotropy to yield a positive-definite value. For Na1 at 600 and 700 °C, anharmonic displacement was observed. Terms up to the fourth order were tested and only kept in refinement if they were significant ([C_{44} ≥ 3[C_{44}]] and led to a significant drop in R values. Thus, two additional unique parameters C_{111}(Na1) and C_{133}(Na1) were refined. Towards the end of refinement, z(Na1) was found to be within a 3σ interval of 0.1 (coplanarity of sodium ions) and was fixed at this value, yielding slightly better residuals despite one free parameter less.

Structure graphics were produced using Diamond 4.5 and VESTA.¹⁴,¹⁵ Refinement results are summarized in Table 2. Diffraction datasets are available in the ILL repository.¹⁶ CSD 1942255 to 1942259 contain the supplementary crystallographic data for this paper; these data can be obtained free of charge from FIZ Karlsruhe via https://www.ccdc.cam.ac.uk/structures.

**Procrystal void analysis**

Procrystal voids in the TiS_2^- x^- substructures were analysed using CrystalExplorer 17.5 with default options.¹⁷ Models were based on the revised room-temperature structure of Na_{0.8}TiS_2·3R_4 and
structures of NaTIS₂–2H₂, NaTIS₂–3R₂, and Na₀.₃TIS₂–6R from the ICSD. Lower bounds for contiguous void networks were found by successively altering the procrystal density in steps of Δρ₀₂₆ = 0.0001 a.u.

MEM reconstruction

Dysnomia 1.0 was used for MEM-reconstruction of SLDs from final structure factors as put out by JANA2006. The unit cell was divided into 96 × 96 × 588 voxels. Starting from a uniform intensity prior, the limited-memory Broyden–Fletcher–Goldfarb–Shanno (L-BFGS) algorithm was employed with uncertainties augmented by E = 0.5 and relative weights set to λ₂ = 1, λ₀ = 0 for n ≥ 4 to avoid overfitting. Final R_p/wR_p were 0.0342/0.0305 and 0.0376/0.0328 for data acquired at 600 and 700 °C, respectively.

OPP calculation

Sodium OPPs were calculated from PDFs as well as from MEM-reconstructed SLDs using CalcOPP 2.0.1. In the latter case, the maximal positive SLD found within the sodium layer was set to a potential energy of V = 0. For error estimation on PDF-derived data, the Monte-Carlo routines implemented in JANA2006 were employed (10 000 iterations, final accuracy < 1%).

Results and discussion

Synthesis and structure

The intercalates NaₓTIS₂ can be synthesized from the binary sulphides in a H₂S feed. At 700 °C, ca. 3–4% of H₂S decompose into H₂ and Sₓ. This creates a mildly reductive atmosphere that acts according to eqn (3).

\[ xNa₂S + 2TIS₂ + xH₂ \rightarrow 2NaₓTIS₂ + xH₂S \]  \hspace{1cm} (3)

Compounds of this type are very hygroscopic and may reversibly co-intercalate up to two equivalents of water per formula unit without degradation. Elemental analyses show that even short handling in air is enough to effect absorption of ca. half an equivalent of water (no significant difference between stored samples with and without neutron diffractometry was observed). At the elevated temperatures examined herein, however, the samples dehydrate completely.

During refinement of the initial model derived from Na₀.₃₅TIS₂–3R₁ against ND data, occupation of a second sodium position at \( \frac{2}{3} \frac{1}{2} \frac{1}{2} \) (mismatch Δz = 0.01), strong correlation of the sulphide ions' z coordinates, and refinement instability occurred. A closer look at the structures in the hitherto assigned space group R₃m and its supergroup R₃m reveals the following differences: the two unique sulphide positions with \( z = 0.39 \) and 0.60 in the former correspond to a single one with \( z = 0.40 \) in the latter (mismatch Δz = 0.01). The two nearly equally occupied sodium positions with \( z = 0.17 \) correspond to a single position that has to be at \( z = \frac{1}{4} \) to warrant a coplanar arrangement (mismatch Δz = 0.0).

We found no hint at symmetry lowering at any temperature. While Na₀.₃₅TIS₂ as synthesised by Rouxel et al. might represent an ordered variant, we assume a case of missed symmetry because of the following reasons: due to the perfect overlap of all reflections in R₃m that are symmetry equivalent in R₃m, only

| Formula | Naₓ₃TIS₂ |
|---------|---------|
| Mᵣ/g mol⁻¹ | 123.48 |
| θ°C | 18 |
| Space group | R₃m |
| a/A | 3.43840(8) |
| c/A | 21.0431(6) |
| Va⁻³ | 215.453(14) |
| Z | 3 |
| Meas./obs. reflections | 73/69 |
| R_p | 0.0268 |
| wR_p | 0.0359 |
| R_ex | 0.0176 |
| R_F | 0.0333 |
| R_l | 0.0520 |

\( ^a I > 3σ(f). \) \( ^b w = 1/|σ²(f) + (0.01f)²]. \)
ever-so-subtle intensity differences distinguish one from the other (see Fig. S2†). For 3R1-isotypic K0.6VS2, Bronsema and Wiegers recognised the same problem and could not resolve it via X-ray powder diffraction."44 Furthermore, both sodium positions in R3m are crystal-chemically nearly identical and should not give rise to ordering. We thus conclude that the published structure is to be revised‡.

Temperature-dependent XRD shows that, on heating, a previously stored Na0.5TiS2-3R1 sample loses co-intercalated water up to 100°C (see Fig. 3, characteristic reflection at ca. 10°). No phase change akin to the 1T–3R transformation§ in the lighter homologue Li0.7TiS2 occurs.45 Further, both sodium positions in R3m are crystal-chemically nearly identical and should not give rise to ordering. We thus conclude that the published structure is to be revised‡.

Cell dimensions lie in the range expected for x = 0.5 at ambient temperature and evolve in a roughly linear fashion during heating (cf. Fig. S7†). The latter also holds for the equivalent displacement parameters of the titanium and sulphide ions, whereas the one of the sodium ion starts at a value five to six times as large and grows disproportionately (see Fig. S8 and S9†). This is due to supposedly static disorder at room temperature and the onset of anharmonic dynamics (i.e., ion diffusion) between 300 and 600°C.

Sodium diffusion pathways

We will describe the methods used herein very briefly. A more detailed overview has already been published.47 The pathways refer to thermally activated ion diffusion at temperatures high enough to warrant ample ion mobility.

Procrystal void analysis. The procrystal void analysis is a fast tool to find the most probable pathways for ion diffusion and to compare compounds, of which only the structure is known.48 Leaving out the mobile (in this case, sodium) ions, the method constructs the remaining static framework as a superposition of diffuse spherical electron densities centred at the atomic positions—the so-called procrystal density ρ_pro. The latter contains voids that are characterised by an isovalue of a surface enclosing them: the lower the value, the “emptier” the void, the better its suitability for containing/conducting an ion.

The low and comparable isovalues, at which the voids in the polytypes 2H, 3R1 (our revised structure), and 6R form contiguous paths, indicate that they are similarly well-suited sodium-ion conductors (see Fig. 4). The framework voids are located at the partly occupied sodium positions and connect to a honeycomb-like conduction pattern. The voids in the 3R2

![Fig. 3](image)

Fig. 3 Heat map of uncorrected temperature-dependent X-ray diffractograms of Na0.5TiS2-3R1 in dinitrogen atmosphere. The colouring is based on raw counts; the scale is logarithmic.

† The previous assignment of the structure type “CuCrSe2–AgCrSe2(R3m)” to Na0.5TiS2-3R1 complicated a comparison to other structures: several isostructural but alloconfigurational structures were assigned this ill-defined type. It was split correctly in the course of this work (see Table S2†), the result being incorporated in the ICSD since May 2019.45

§ The compounds Li,TiS2-3R are isotypic to the high-alkali polytype Na,TiS2-3R1, not Na,TiS2-3R2.

![Fig. 4](image)

Fig. 4 Procrystal void surfaces (blue) in the sodium layers of the TiS2x⁻ frameworks for different polytypes. Atoms are plotted with covalent radii (grey: titanium, yellow: sulphide ions). The depicted isovalue is the minimum for a contiguous surface.

‡ Caveat: There had been a transcription error in the former ICSD record that was corrected in 2018.
structure also arrange in a honeycomb-like fashion and give no hint at any reason for an ordered occupation of half of them (like the current structural model suggests). The voids connect at a significantly higher isovalue to a carpet-like pattern, which suggests that this polytype is a worse and more isotropic sodium-ion conductor.

**Probability-density function.** The PDF of an atom at a point in space describes the probability of finding said atom displaced from its equilibrium position to said point. In the case of harmonic displacement, the depiction of the PDF as isosurface leads to the well-known spheres (isotropic) or ellipsoids (anisotropic). In the anharmonic case, however, forms that are more complex ensue. Because of this, a superposition of the PDFs of several atoms, called a joint probability-density function (JPDF), allows the visualisation of diffusion pathways.

In the case at hand, the sodium ions are displaced along linear pathways between adjacent sodium positions at both relevant temperatures (cf. the similar isosurfaces in Fig. 5). In this way, the honeycomb-like conduction pattern predicted by procystal void analysis is indeed experimentally reproduced.

**MEM-reconstructed scattering-length density.** To rule out bias, we have also inspected maps of the MEM-reconstructed SLD, which do not rely on modelling atomic displacement. In comparison to maps directly Fourier-transformed from the structure factors, they are deprived of artefacts and show a flatter distribution of noise. The maps show that our PDF model is indeed accurate and reproduces even finer features of the positive sodium-associated SLD (see Fig. 6).

In the lighter but more alkali-rich congeners Li\textsubscript{x}TiS\textsubscript{2}3R (x = 0.7, 0.9), preferred ion pathways were found between adjacent lithium positions.\textsuperscript{49} Because of their ordering, however, they correspond to second-neighbours in disordered Na\textsubscript{0.5}TiS\textsubscript{2}3R\textsubscript{1}.\textsuperscript{3} As a result, the pathways are not directly comparable. It is noteworthy that, in contrast to Li\textsubscript{0.7}TiS\textsubscript{2}3R, we have not found a significant deviation from coplanarity. We attribute this to the differences in coordination polyhedra (trigonal antiprism vs. prism) and spatial arrangement (interlayer spacing larger by \(\Delta d \approx 0.96 \text{\AA}\), while ionic diameter only larger by \(2\Delta r = 0.52 \text{\AA}\) (ref. 50) for the sodium compound).

Migration barriers

For the estimation of the migration energy barrier associated with the found path, we calculated the OPP, which represents the energy landscape experienced by a mobile sodium ion (approximated as an Einstein oscillator that is subject to Boltzmann statistics at the classical limit). The activation energy of the migration (not comprising the defect formation energy) is the OPP at the bottleneck position, i.e., the position of lowest probability along the pathway.

The calculation of OPPs from ND-derived PDFs is fairly established,\textsuperscript{51} but direct derivations from MEM-reconstructed SLDs without any further refinement are scarce.\textsuperscript{26} The latter is only warranted if an SLD shares basic characteristics of a probability density. Mathematically speaking, a PDF has to be non-negative, Lebesgue-integrable, and normalized to an integral of unity over the whole space. To give a reason for using the direct approach nonetheless, the following points address these issues not in a rigorous but hand-waving way:

- The OPP calculation deals with ratios of input values. Only the non-negative ones are taken into account.
- Numerically sampled SLDs are Lebesgue-integrable.
- For ratios of input values, normalization is unnecessary (the normalizing constant cancels).

Furthermore, the adequacy of the results is ensured by only taking the SLD solely caused by sodium ions (i.e., within the sodium layers) into account.

In agreement with crystal-chemical assumptions, we found the bottleneck of migration at \(\frac{1}{4}, \frac{1}{4}, \frac{1}{4}\)—in the middle of a straight line between adjacent sodium positions. The OPPs at this point are summarized in Table 3. Generally, the activation energy does not depend on the temperature as long as only a single
migration mechanism is concerned and the structure is stable. For this method, however, this holds only if diffusion is fully activated because the migrating ions themselves are the probe for the potential. This is why slightly lower activation barriers are often found for higher temperatures—as is the case here for the central values—and are thought to be more reliable. Nevertheless, the PDF-derived energies are equal within 1σ. Currently, there is no computational way to estimate the errors on SLD-derived values. Judging by the uncertainties of the PDF-derived barriers, however, the central values are significantly lower, but of the same magnitude. As this is the first study employing both methods, it remains yet unknown if the deviation is systematic. It is, e.g., possible that a restricted set of parameters models parts of the SLD imperfectly (thereby modelling a too low probability density resulting in a too high OPP) or that the SLD-derived OPP carries a considerable error.

The topography of the OPP landscape, on the other hand, is similar irrespective of temperature or calculation method (see Fig. 7). As temperature rises, the potential becomes flatter and gives access to a larger area (cf. Fig. 7a/b), but the basic shape is preserved. As the sodium PDF reproduces the features of the SLD (vide supra), the landscapes derived from both resemble each other (cf. Fig. 7c/d). The smaller volume enclosed by the same isosurface, however, shows that the SLD-derived OPP is steeper, therefore stemming from a more localised distribution.

The experimental barriers are lower than computed for defective NaTiS$_2$-$3R$ (0.19 eV) and measured for Li$_{0.7}$TiS$_2$-$3R$ (ca. 0.5 eV). Procrystal void analysis already led us to expect that these homeotypic (by alkali occupation) structures be detrimental to alkali-ion conduction (vide supra). In addition, the lower number of vacancies (higher alkali content) in the latter hinders migration. For Na$_{0.5}$TiS$_2$-$3R_1$, low activation barriers corroborate its role as a good sodium-ion conductor. Although a model all-solid-state sodium-ion battery using a Na$_{0.5}$TiS$_2$ electrode has been realized, problems with reactivity, phase transformations upon de-/intercalation, and mediocre cyclability seem to prohibit its effective use therein.

**Remarks on “Na$_{0.9}$TiS$_2$-2H”**

Our endeavour to explore also the more sodium-rich Na$_{0.9}$TiS$_2$-2H was hindered by synthesis problems. While XRD for phase identification and ND seemed to indicate a single-phase product of the desired polytype, closer inspection (see Fig. S3f) revealed at least two similar phases, possibly with a more complex (OD) structure. Because of the broader reflections in ND, we found no way of separating the contributions of different phases. A refined (unphysical) single-phase model averaging over all phases was at odds with elemental analysis: The former found only ca. 0.5 sodium ions per formula unit at the crystallographic positions (no significant positive residual SLD) while the latter conformed to the target composition.

Furthermore, we found a strong preferred orientation in (001) that agrees well with the plate-like visual appearance of the crystallites in the sample. In a temperature-dependent XRD experiment, we observed the 2H-like structure over the whole stability range. No signs of a transformation into a supposedly stable 3R$_2$-like structure showed above 300 °C, even when heating for a longer time.

**Conclusions**

We synthesised the intercalate Na$_{0.5}$TiS$_2$-$3R_2$ from sodium and titanium sulphide in a mildly reducing H$_2$S atmosphere. Our ND and XRD experiments showed that its structure, as reported in the literature, has to be revised: Na$_{0.5}$TiS$_2$-$3R_1$ assumes the space-group type $R3m$ (not $R3m$) with disordered (not ordered) sodium ions and one (not two) crystallographically independent sulphide position. Up to ca. 100 °C, the compound reversibly co-intercalates half an equivalent of water. It is stable with respect to phase transformations up to its decomposition temperature.

Sodium-ion diffusion manifests in anharmonic displacement at 600 and 700 °C. In agreement with expectations from topological analysis of the static TiS$_2$ framework, maps of the MEM-reconstructed SLD and modelled PDFs show a honeycomb-like conduction pattern with linear almost in-plane

| Table 3 Activation energies (in eV) of sodium-ion migration in Na$_{0.5}$TiS$_2$-$3R_1$ |
| --- |
| Calculation | At 600 °C | At 700 °C |
| From displacement-derived PDF | 0.114(7) | 0.108(8) |
| From MEM-reconstructed SLD | 0.086 | 0.070 |
pathways between adjacent (partly occupied) sodium positions. The bottlenecks of migration are found at the midpoint of said pathways. The associated OPPs at 700 °C indicate low activation barriers of 0.108(8) eV or even 0.070 eV (derived from PDF or SLD, respectively).

Unfortunately, we could not complement these findings with data for Na$_{0.9}$TiS$_2$·2H. We presume that our synthesis method led to segregation into at least two 2H-like phases—one with lower, one with higher sodium content. Possibly, these are closely related to Na$_2$TiS$_3$ with its OD structure. Bouwmeester et al. had already found such by-phases (re-)forming during high- to mid-temperature syntheses ex elementis with high sodium load. The potential of arranging

- pure sodium layers (e.g., in Na$_x$TiS$_3$), their defective (e.g., in Na$_x$TiS$_2$ with $x < 1$), titanium-mixed (e.g., in Na$_x$Ti$_2$S$_3$), or even as-well-as variants (e.g., in Na$_x$Ti$_{1-x}$S$_2$ with $x < 1 – 3d$)
- with their sodium, titanium, and vacancy positions ordered or unordered within one layer
- in an ordered or unordered stacking sequence

accounts for compelling structural complexity in layered sodium titanium sulphides. This is particularly pronounced at the fringes of very high or very low alkali content and makes further exploration, especially of the interplay between sodium mobility and structural disorder, worthwhile.

Conflicts of interest

There are no conflicts to declare.

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