**Tough and Variable-Band-Gap Photonic Hydrogel Displaying Programmable Angle-Dependent Colors**

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**ABSTRACT:** One-dimensional photonic crystals or multilayer films produce colors that change depending on viewing and light illumination angles because of the periodic refractive index variation in alternating layers that satisfy Bragg’s law. Recently, we have developed multilayered photonic hydrogels of two distinct bulk geometries that possess an alternating structure of a rigid polymeric lamellar bilayer and a ductile polyacrylamide (PAAm) matrix. In this paper, we focus on fabrication of composite gels with variable photonic band gaps by controlling the PAAm layer thickness. We report programmable angle-dependent and angle-independent structural colors produced by composite hydrogels, which is achieved by varying bulk and internal geometries. In the sheet geometry, where the lamellae are aligned parallel to the sheet surface, the photonic gel sheet exhibits strong angle-dependent colors. On the other hand, when lamellae are coaxially aligned in a cylindrical geometry, the gel rod exhibits an angle-independent color, in sharp contrast with the gel sheet. Rocking curves have been constructed to justify the diverse angle-dependent behavior of various geometries. Despite varying the bulk geometry, the tunable photonic gels exhibit strong mechanical performances and toughness. The distinct angle dependence of these tough photonic materials with variable band gaps could benefit light modulation in displays and sensor technologies.

**INTRODUCTION**

Photonic hydrogels, which possess a three-dimensional polymer network structure with periodic variations of refractive index, are potential candidates for soft and tactile displays, sensor systems, packaging, and advertising. The ordering present in these soft hydrogels causes them to exhibit a “structural color” that can be reversibly tuned by varying the periodic spacing through external stimuli such as stress, temperature, pH, and solvent exchange or by varying the refractive index of the material. Therefore, photonic hydrogels are promising materials for displays and sensors through utilizing stimuli-responsive structural colors rather than requiring colored filters or additional optical elements. The structural color of materials that possess one-dimensional photonic crystal structures or multilayer films shows distinct variations depending on viewing and light illumination angles. In contrast, amorphous photonic structural arrays are responsible for viewing-angle-independent colors. Angle-dependent continuous color filtering allows the multicolor appearance of see-through organic photovoltaics and plasmonic chips for the construction of reflection-based spatial light modulators. On the other hand, changes in color associated with viewing or illumination angle variation can also pose major challenges for the development of full-color displays with wide viewing angles. Therefore, both tunable angle-dependent and angle-independent colors in photonic materials are high in demand for optical devices. Researchers, to date, have shown that materials having a short-range order in periodic structures display angle-independent colors originating from a wavelength-specific constructive interference in the visible region. Structurally colored materials that exhibit angle-dependent colors most likely possess long-range periodic structures. Takeoka reported angle-dependent colors by constructing colloidal crystalline arrays and angle-independent colors by assembling colloidal amorphous arrays. To the best of our knowledge, tuning between angle-dependent and angle-independent colors by modifying the photonic nanostructure in line with the bulk geometry has rarely been reported.

On the basis of long-range periodic lamellar structures, we have recently developed a tough photonic hydrogel that exhibits tunable colors through various stimuli including stress, temperature, and pH. The photonic hydrogel consists of
an alternate packing of rigid polymeric bilayers [self-assembly of amphiphilic poly(dodecyl glyceryl itaconate) (PDGI)] and ductile polyacrylamide (PAAm) layers (Figure 1). Long-range periodic photonic structures have been fabricated by shear-induced orientation of dodecyl glyceryl itaconate (DGI) bilayer domains during preparation. A gel precursor solution containing DGI monomers undergoes self-assembly to form lamellar bilayers that can be aligned along the shear direction. Given this ability, long-range ordering of lamellar bilayers can be controlled by the geometry of the reaction cell, which determines the shear flow of the precursor solution during injection. When the precursor solution is injected into a rectangular reaction cell consisting of two parallel glass plates, lamellar bilayers are aligned parallel to the glass wall. Upon UV curing, hydrogel sheets are formed, where lamellar bilayers uniaxially appeared along the sheet surface and trapped between PAAm matrices (Figure 1, left). On the other hand, rod-shaped hydrogels, in which bilayers form a concentric multilamellar packing, have been achieved by applying suction to a precursor solution into a cylindrical polyethylene tube (Figure 1, right).

Herein, we report the fabrication of composite hydrogels (PDGI/PAAm) with variable photonic band gaps and the resulting angle-dependent or angle-independent structural colors based on the specific photonic nanostructure and bulk gel geometry. Gel sheets and rods of varying photonic band gaps within the visible spectrum were fabricated by controlling the swelling degree of the PAAm matrix. Photonic gel rods exhibited angle-independent colors, in sharp contrast to gel sheets, which showed strong angle dependence. The contrasting angle dependence of both the gels is characterized by constructing rocking curves and comparing their lamellar geometries. Despite various bulk geometries with a distinct lamellar packing, both the gels are soft and ductile, exhibiting high mechanical strength and toughness. Understanding the relationship between the photonic nanostructure and the angle dependence of these soft and tough hydrogels could impact the material design for light modulation, visualization, optical sensing, and display technologies.

■ RESULTS AND DISCUSSION

Photonic hydrogel sheets possess several thousands of rigid PDGI lamellar bilayers stacked periodically into a soft and ductile PAAm matrix. PDGI lamellae are entrapped in a chemically cross-linked PAAm network by one-step polymerization in the presence of a small amount of a chemical crosslinker. In the as-prepared state, gel sheets [0.1 M DGI, 2.0 M acrylamide (AAm), and 0.1 mol % N,N′-methylenebis(acrylamide) (MBAA) of AAm] are almost transparent and exhibit a pale violet color (Figure S1). However, visible colors appeared depending on the swelling degree in pure water. PDGI bilayers are impermeable to water and other hydrophilic molecular species. Because of the uniaxial orientation of the PDGI bilayers, the water-permeable PAAm network can swell freely in the direction perpendicular to the bilayers and the bulk thickness of the gel sheet almost doubled. The swelling is related to the increase in the PAAm layer thickness and a corresponding increase in the inter-bilayer distance, which leads to a red shift in the gel color. The thickness of a single PDGI bilayer (d_{PDGI}) was reported in the literature previously as 4.7 nm. Therefore, the thickness of each PAAm layer (d_{PAAm}) can be estimated from d_{PAAm} of the water-swollen gel sheet, which exhibits an orange-red color, is estimated as 250 nm. At room temperature, the single PDGI bilayer membrane is rigid with a modulus on the order of several megapascals and the PAAm matrix is soft with a modulus of few kilopascals.

The refractive index of a PDGI bilayer membrane is considered to be that of a lipid bilayer, which is reported in the range of 1.42–1.45. The refractive index of a PAAm layer is considered to be 1.33 (the refractive index of water) as it contains mostly water (98%). Therefore, the PDGI/PAAm composite gel exhibits visible colors at various wavelength
bands because of the periodic refractive index variation of alternating layers that satisfy the Bragg’s law for diffraction of visible light.

The periodic lamellar spacing, $d$, of the gel sheets and correspondingly their color can be tuned by varying the DGI concentration during preparation, mechanical stress, and so forth. In this work, $d$ was varied by regulating the PAAm layer thickness. This was achieved by controlling the swelling degree of the PAAm matrix. The PDGI/PAAm gel sheets were prepared at various cross-linking concentrations using MBAA as a cross-linker in the PAAm network. The photographs of the gel sheets prepared with various MBAA concentrations are shown in Figure 2a. With increasing MBAA concentration, a blue shift is observed in accordance with the corresponding reflection spectra, as shown in Figure 2b. The wavelength of maximum intensity ($\lambda_{\text{max}}$) and interlamellar distance ($d$) decrease accordingly (Figure 2c). Therefore, a shift in the gel color is attributed to the decrease in $d$. At a high cross-linking density, the swelling degree of the PAAm matrix was reduced, which resulted in a decrease in $d$. The bulk swelling ratio along the thickness of the gel ($T/T_0$) also decreased in accordance with the $d$-ratio ($d/d_0$) (Figure 2d), where $T_0$ and $d_0$ and $T$ and $d$ are the reference states (as-prepared) and swollen states, respectively, for gels with various cross-linking concentrations.

The gel rods were prepared with compositions identical to those of the gel sheets (0.1 M DGI, 2.0 M AAm, and 0.1 mol % MBAA of AAm), resulting in an almost transparent gel that resulted in a decrease in the swelling degree of the PAAm matrix. The PDGI/PAAm gel sheets were positioned in the polarizer direction (Figure 4b(iii)). However, the images over the whole sample when the long axis of the rod was further revealed by the POM images taken perpendicular to the top surface of the gel sheets. The top-view images, which show solid magenta colors for all sample angles, indicate no in-plane alignment, further justifying the in-plane flat lamellae (Figure 4a(iv-vi)).

The rod-shaped gel was also observed under cross-polarizers from the top as well as from a cross-sectional view (Figure 4b). POM images, which were taken from the cross-section of the rod in the presence (Figure 4b(i)) and absence (Figure 4b(ii)) of a tint plate, show a Maltese cross, which clearly indicates a concentric alignment of bilayers along the periphery of the rod. Large-scale concentric cylindrical lamellae of the gel rods were further revealed by the POM images taken perpendicular to the long axis of the rod. POM produces completely dark images over the whole sample when the long axis of the rod was positioned in the polarizer direction (Figure 4b(iii)). However, at $-45^\circ$ and $+45^\circ$ rotations, orange and blue birefringences were observed, respectively, together with a thin line of magenta color in the center along the long axis (Figure 4b(iv-vi)). These birefringence micrographs clearly justify the

Figure 2. (a) Photographs of the gel sheets prepared at various cross-linker concentrations (MBAA) show a blue shift with an elevated MBAA content. (b) Reflection spectra of the gels are measured at various cross-linker concentrations. Both incident and detection angles ($\theta$) were set at 60° for the spectra measurements. (c) Wavelength of maximum intensity ($\lambda_{\text{max}}$) and interlamellar distance ($d$) are plotted as a function of cross-linker concentration. $d$ was calculated from $\lambda_{\text{max}}$ using Bragg’s law, $2nd \sin \theta = m\lambda$, where $n$, $d$, $\theta$, $m$, and $\lambda$ are the refractive index of water, periodic lamellar spacing, Bragg’s angle, diffraction order, and wavelength of diffracted light, respectively. (d) Bulk swelling ratio of gel thickness ($T/T_0$) and $d$-ratio ($d/d_0$) are shown as a function of cross-linker concentration, where $T_0$ and $d_0$ and $T$ and $d$ stand for the reference state (as-prepared) and swollen state, respectively, of gels with various cross-linking densities. Cross-linker concentrations are expressed as mol % with respect to AAm molar concentrations. The scale bar is $\sim$5 mm.
existence of large-scale cylindrical lamellae with a concentric alignment in the gel rods.

Although the gel sheets and rods possess a distinct bulk geometry as well as internal nanostructured packing, they exhibit comparable mechanical strength and toughness (Figure 5). Both the gels possess a modulus of ~100 kPa and a work of extension on the order of several MJ m$^{-3}$.

To observe the angle dependence of the structural color, real-time movies and photographs of both gel sheets (0.1 mol % MBAA) and rods in the red photonic band (0.025 mol % MBAA) were taken at different viewing angles (Figure 6a,b and Movies S1 and S2). The gel sheet exhibits an orange-red color at a viewing angle of 0°. The color changes from red to blue with increasing viewing angle on both directions (±30, ±45, and ±60°) (Figure 6a). This angle-dependent behavior is similar to some natural phenomena such as soap bubbles, beetle’s exoskeletons, and bird’s feathers.44−46 The gel rods, which possess an orange-red color at 0°, do not undergo any noticeable shift in color by changing the viewing angle in both directions, which is in sharp contrast to the sheets (Figure 6b and Movie S2). It should be noted here that we chose gel rods of different preparation conditions (0.025 mol % MBAA) to

Figure 3. (a) Photographs of gel rods prepared at various cross-linker concentrations (MBAA) show a blue shift at a higher MBAA content. (b) Gel rod bulk swelling ratios in the diameter direction ($D/D_0$) decrease with increasing cross-linker concentration. The scale bar is ~2 mm.

Figure 4. POM images of gel sheets (a) and rods (b). All of the images are taken under cross-polarizers in the presence of a tint plate (530 nm) [except for b(ii)]. The birefringence of the samples is assigned to the monodomain bilayers and elongated PAAm chains because of the anisotropic swelling. [a(i−iii)] Cross-sectional images of a gel sheet at 0° to the polarizer (i) appear magenta but orange at −45° (ii) and blue at +45° (iii), indicating uniaxially aligned lamellae parallel to the top surface of the gel. [a(iv−vi)] Top-view POM images of the gel sheet show only magenta color at all angles when the sample stage is rotated, which indicates no in-plane sheet lamellae. [b(i)] Images taken from the cross-section of the gel rod in the presence (i) and absence (ii) of a tint plate show a Maltese cross, which clearly indicates the concentric alignment of lamellae to form a multicylindrical packing. [b(iii−v)] Images taken perpendicularly to the long axis of the rod (top view) are completely magenta (iii) over the whole sample when placed at 0° to the polarizer but orange (iv) and blue (v), separated by a solid magenta line, when placed at −45° and +45°, respectively. This further indicates a large-scale concentric cylindrical alignment of lamellae in the gel rods. The scale bars are 1.0 mm.

Figure 5. Nominal stress vs strain curves of a gel sheet and a gel rod indicate comparable levels of mechanical properties. The modulus of both the gels is ~100 kPa. Compositions of these two gels at preparation are identical (0.1 M DGI, 2.0 M AAm, and 0.1 mol % MBAA of AAm). The deformation velocity was 200 mm/min.
Achieve a color identical to that of the gel sheet for the angle-dependent study. Although long-range, flat multilayer structures produce high angle dependence in the case of gel sheets, rods with cylindrical lamellar structures exhibit negligible angle dependence, which is analogous to the structural color in *Morpho* butterflies. 24

The reflection spectra of both gel sheets and rods were measured at various viewing angles using an angle-measuring reflection apparatus. During the measurement, incident (θ_in) and reflection (θ_out) angles were identical, where θ_in = θ_out (=90 °−θ_view), where θ_view is the viewing angle. The reflection spectra of the gel sheets showed a blue shift in the peak wavelength (Figure 6c), whereas the peak wavelength of the rods does not change with increasing θ_view (Figure 6d), which is in consistence with their appearance. The wavelength at maximum intensity (λ_max) for both the gels was plotted as a function of θ_view and is demonstrated in Figure 6e. A large shift in the peak wavelength (∼120 nm) is observed for the gel sheet by changing the viewing angle from 10 to 55 °. In contrast, no significant change in λ_max for the gel rod is observed. However, the peak intensity of the gel rod is much lower than that of the sheet geometry. The angle dependence of the gels is related to Bragg’s law, 2nd sin θ = ml, where n, d, θ, m, and λ are the refractive index of water, periodic lamellar spacing, Bragg’s angle, diffraction order, and wavelength of diffracted light, respectively. 47–49 As the sheet geometry possesses flat lamellae, the Bragg’s angle (θ) is equivalent to θ_in or (90 °−θ_view). According to Bragg’s law, lower wavelength light is reflected at a smaller Bragg’s angle, which is why a blue shift occurred in the gel sheet when observed at a higher viewing angle. On the other hand, because the rod geometry possesses cylindrical lamellae, incident/ reflected light impinges on the circular lamellar surface at various angles. Although the viewing angle is changed spatially, the true angle of incidence and reflection with respect to the sample position remains unchanged. Therefore, the phenomena of Bragg’s diffraction do not vary upon rotation of the cylindrical lamellar gel. Regardless of the sample position or viewing angle, incidence and reflection always appeared at an angle of 90 ° on cylindrical lamellae for the gel rods.

To quantitatively characterize the angle dependence, we further construct rocking curves where the incident (θ_in) and detected (θ_out) beams with respect to the initial position of samples were set in such a way that the angle between the incident and detected beams (ω) remains fixed upon rocking the sample (Figure 7a,b). Keeping ω = 50 °, both the gel samples were rocked at various angles (ψ).

The reflection spectra of both gels taken at various rocking angles are shown in Figure 7c,d. The reflection peak intensity of the gel sheet falls sharply by rocking the sample at a small angle of only ∼10 °, whereas a negligible change in the peak intensity is noticed for the rod while rocking even at an angle of 30 °. The rocking curves, intensity ratio (I/I_0) versus rocking angle, for both gels are plotted, where I and I_0 are the reflection peak intensities at various rocking angles and at zero rocking angle, respectively (Figure 7e). A sharp rocking impact is apparent for the gel sheets, whereas a broad impact can be seen for the rods. The maximum intensity observed at 0 rocking angle for the gel sheet confirms that lamellar bilayers are mostly aligned parallel to the top surface. In the case of the gel rods, the angle of the incident beam and detector remained unchanged with respect to the sample position regardless of the rocking angle, indicative of angle-independent structural colors. With the gel...
sheet, the incident and detection angles with respect to the sample position always change with varying rocking angle, demonstrating highly angle-dependent structural colors. Similar results were obtained when the angle, \( \omega \), between the incident and reflected beams is varied for both gels (Figure S2). Full width at half-maximum (fwhm) for both gels is also plotted as a function of rocking angle in Figure 7f. The bilayer undulation in the gel sheet may affect its rocking intensity distribution, whereas there is negligible influence on rocking for the gel rod. Therefore, the rocking curves clearly confirmed the distinct angle dependence of both gels.

The distinct angle-dependent behavior has been previously demonstrated by constructing colloidal crystalline arrays or by assembling colloidal amorphous arrays. These photonic systems are isotropic on the macroscale and have limited applications because their structure rarely influences their mechanical performances. Although rolled-up multilayered photonic fibers show moderately improved mechanical functions, they exhibit only angle-independent colors. On the other hand, our photonic gel system is anisotropic (on both the macro- and microscales) and exhibits excellent mechanical functions as well as color tunability and promising optical occurrences by modifying the external geometry along with the internal photonic nanostructure. Moreover, our photonic system could guide us to understand the unique optical occurrences in nature as well as various biological organisms, such as soap bubbles, beetles, fishes, cuticles, butterflies, and some types of bird feathers.

**CONCLUSIONS**

Photonic hydrogels with tunable angle-dependent structural colors have been developed, which is achieved by varying the bulk geometry and their photonic nanostructures. Hydrogel sheets with uniform lamellar domains exhibit strong angle-dependent structural colors, whereas gel rods with concentric cylindrical lamellae show angle-independent colors, in sharp contrast to the sheet geometry. Rocking curves have been constructed to justify the diverse angle dependence of lamellar photonic hydrogels with various bulk geometries. Both gel sheets and rods with variable photonic band gaps have been fabricated by varying the swelling degree of the PAAm layer. Despite varying the bulk geometry, softness, and ductility, both gels exhibit high mechanical performances and toughness. Owing to the soft and tough nature of these materials, stimulus-responsive photonic band gap, and distinct angle dependence, these promising photonic nanocomposite materials could be potentially useful in display technologies and to uncover unique optical occurrences in nature.

**EXPERIMENTAL SECTION**

**Materials.** An amphiphilic monomer, DGI (\( n-C_{12}H_{25}−OCOCH_3(=CH_2)COOCH_2CH(OH)CH_2OH \)), was synthesized following the procedure described earlier. After completion of two-step reactions, the crude product was purified at least twice by a silica gel column (silica gel 60 N, Kanto Chemical Co., Inc.). The DGI fraction was eluted with a hexane/ethyl acetate mixture (1:1 by volume) and was further...
purified twice by recrystallization from an acetone/hexane mixture (1:1 by weight). MBAA was recrystallized from ethanol, AAm was recrystallized from chloroform, and Irgacure 2959 and sodium dodecyl sulfate (SDS) were used as received. Millipore deionized water was used to prepare the monomer solutions and for the swelling of the gel.

Preparation of Gel Sheets and Rods. PDGI/PAAm gel sheets and rods were prepared by simultaneous free-radical polymerization from an aqueous solution of 0.1 M DGI, 0.025 mol % SDS of DGI, 2 M AAm, 0.025−5.0 mol % MBAA of AAm as a cross-linker, and 2 mM Irgacure as an initiator.34,36 The precursor solution was prepared by mixing all components which contained random microdomains of self-assembled DGI lamellar bilayers dispersed in an AAm solution. Gel sheets with flat lamellar bilayer structures parallel to the sheet surface were fabricated by applying shear flow of an equilibrated precursor solution between two parallel glass plates. Prior to polymerization, the precursor solutions were injected in a rectangular-shaped glass mold (100 × 10 × 0.5 mm³) at a shear rate of ~200 s⁻¹, where the two parallel glass plates were separated by a 0.5 mm thick silicon spacer (Figure 1, left). Because of the strong shear flow between the two parallel glass plates, random lamellar bilayer domains were aligned in the direction parallel to the glass substrate. After 8 h of UV polymerization, uniform, flat lamellae were stacked periodically and entrapped in a chemically cross-linked PAAm matrix to create the hydrogel sheet. On the other hand, PDGI/PAAm gel rods were fabricated by using a polyethylene tube (1.0 mm diameter and 200 mm long) as a reaction cell to apply strong shear flow along the axial direction of the tube (Figure 1, right). The gel precursor solution was prepared in the same manner as above. One end of the tube was immersed in the precursor solution, and the other end was connected to a syringe. Suction was applied to the precursor solution at a high speed (shear rate > 200 s⁻¹) until the tube was filled. Under strong shear flow, random lamellar domains in the precursor solution were oriented along the wall of the tube to form multicylinder lamellae. The rod structure was stabilized in the PAAm matrix after 8 h of UV polymerization. Both gels were immersed in water to achieve an equilibrium swollen state.

Reflection Spectra Measurements. Variable angle reflection measurement optics (Hamamatsu Photonics K.K., C10027A10687) were used for exposing the gels to white light and detecting the reflected light. A xenon lamp was used as a light source to measure the reflection spectrum. A photonic multichannel analyzer (Hamamatsu Photonics K.K., C100027) was used for analyzing the detected signal. To observe the angle-dependent color, the reflection spectra of the gel samples were measured at various viewing angles (θview) while setting both incident (θin) and reflection (θout) angles to 90°−θview (θin = θout = 90°−θview). Reflection spectra measurements for the rocking curves were performed by rocking the sample in both directions (Figure 7a,b). The incident beam angle (θin) and detection angle (θout) with respect to the initial sample position were set in such a way that the angle between the incident beam and the detector (ω) remained unchanged during rocking. When ω = 50°, θin = θout = 65° at 0 rocking angle (ψ). At ψ = 5°, ω = 50° and θin = 60° while θout = 70°.

Tensile Test. Tensile stress−strain properties of both gel sheets and rods were measured with a commercial test machine (Tensilon RTC-1310A, Orientec Co.). Prior to testing, the gel sheets were cut into a dumbbell-shaped standardized size with a gel cutter (JIS-K6251-7) in such a way that the tensile elongation could be performed parallel to the top surface of the gel, that is, in the direction along the lamellae. The gel rods were clamped as is. The distance between the two clamps was considered as the initial length of the gel rod. Deformation was applied along the long axis of the gel rod. The stretching velocity was 200 mm/min for both gels.

ASSOCIATED CONTENT

Supporting Information
The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsomega.7b01443.

Additional results related to the swelling of the as-prepared gels and reflection spectral analysis for rocking curves (PDF)

Real-time movie taken by a conventional video camera to show the angle-dependent color of the gel sheet (AVI)

Real-time movie taken by a conventional video camera to show the angle-independent color of the gel rod (AVI)

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Notes
The authors declare no competing financial interest.

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