One-Step Hydrothermal Synthesis of P25 @ Few Layered MoS$_2$ Nanosheets toward Enhanced Bi-catalytic Activities: Photocatalysis and Electrocatalysis

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Abstract: P25 loaded few layered molybdenum disulfide (MoS$_2$) nanosheets (P25@MoS$_2$) are successfully synthesized through a facile one-step hydrothermal process. The bi-catalytic activities, i.e., photocatalytic and electrocatalytic activities, of the as-prepared nanomaterials have been investigated. For the as-prepared products, the photocatalytic performances were investigated by degrading simulated pollutant under sunlight irradiation, and the hydrogen evolution reaction evaluated the electrocatalytic performances. The results indicate that P25@MoS$_2$ possesses excellent activities in both photocatalysis and electrocatalysis. The presence of MoS$_2$ broadens the light absorption range of P25 and improves the separation and transformation efficiency of photogenerated carriers, thus improving its photocatalytic performance. The existence of P25 inhibits the aggregation of MoS$_2$ to form more dispersed MoS$_2$ nanosheets with only few layers increasing its active sites. Thereby, the electrocatalytic performance is heightened. The excellent multifunction makes the as-prepared P25@MoS$_2$ a promising material in the fields of environment and energy.

Keywords: P25; MoS$_2$; bi-catalysis; photocatalysis; electrocatalysis

1. Introduction

In recent years, the environmental pollution and energy shortage that have been caused by the extensive use of fossil energy threatened the healthy and survival of human beings. Therefore, the bottleneck problem of energy and environmental pollution, which restricts human development, needs to be urgently solved. More renewable energy and less fossil energy are advantageous for environment and sustainable development. Among many renewable resources, solar energy has become the preferred alternative energy due to its advantages of time-durability and environmental friendliness.

Among the ways of utilizing solar energy, photocatalytic technology is more popular due to its direct use of clean solar energy to solve environmental pollution problems. Many semiconductor nanomaterials, exhibiting an excellent performance of photocatalysis, have been investigated, such as ZnO, CuO, CdSe, etc. [1–4]. Among these semiconductors, titanium dioxide (TiO$_2$) is undoubtedly the most shining one. Although TiO$_2$ has extra-outstanding activity in photocatalysis, it still cannot get rid of the two inherent weaknesses: wide-band-gap and low quantum efficiency [5]. The combination
of two semiconductor materials with a matched energy level is one effective way to solve these two deficiencies of TiO$_2$ [6,7].

In addition, using solar energy to catalyze the production of hydrogen is intensively concerned at present [8–10]. However, the direct use of solar energy for photocatalytic hydrogen production will be strongly affected by weather, and thus affect the stability and sustainability of hydrogen production. Electrocatalytic hydrogen production is a promising means of preparing alternative energy sources, owing to the fact that various renewable energies (wind, waves, tide, etc.) can be converted to electric energy [11]. In this way, various renewable energy can be converted, stored, and transported by using hydrogen as the medium [12]. Many nanomaterials for electrocatalysis have been produced and researched, such as Ni$_2$P [13], CoSe$_2$ [14], C$_3$N$_4$@NG [15], molybdenum disulfide (MoS$_2$) [16], and so on.

Photocatalysis and electrocatalysis, both effective ways, have their own advantages in utilizing renewable energies. Therefore, designing a system that can operate under light and electricity is meaningful. The core of this system is to synthesize a nanomaterial possessing both excellent photocatalysis and excellent electrocatalysis. For this purpose, the composite of commercial P25 and few layered MoS$_2$ nanosheets (P25@MoS$_2$) is synthesized.

P25 is consists of TiO$_2$ nanoparticles with an average size of 25 nanometers, which has extra outstanding activity in photocatalysis, and it has been considered as a reference of photocatalyst [17]. However, P25 has no excellent electrocatalytic performances. MoS$_2$ has excellent electrocatalytic activity, especially in hydrogen evolution [18,19]. It has been reported that MoS$_2$ edge structure resemble nitrogenase active site, and the free energy of MoS$_2$ edge is close to Pt [20]. However, MoS$_2$ has no excellent photocatalytic performance. When considering that P25 has excellent photocatalysis and MoS$_2$ has excellent electrocatalysis, the combination of P25 and MoS$_2$ is expected to possess excellent bi-catalysis- photocatalysis and electrocatalysis.

Zheng et al. synthesized a hierarchical MoS$_2$ nanosheet@TiO$_2$ nanotube array by combining the anodization method with the hydrothermal method for enhanced photocatalysis in degrading rhodamine B (RhB) [21]. Song and co-workers prepared the MoS$_2$/TiO$_2$ hybrids with various interfaces by a three-step hydrothermal method for electrocatalytic hydrogen evolution reaction (HER) [22]. The MoS$_2$/TiO$_2$ etching in H$_2$SO$_4$ had the best performance in all of the samples. Ma et al. reported flower-like MoS$_2$@TiO$_2$ nanohybrids for hydrogen evolution by the two-step hydrothermal method [23]. The results showed that MoS$_2$@TiO$_2$ had improved photocatalysis, and the sample (14.6 wt% MoS$_2$) had an onset overpotential of −340 mV (vs RHE) and a Tafel slope of 81 mV dec$^{-1}$. However, these excellent achievements need a complex process to composite MoS$_2$@TiO$_2$. Moreover, the electrocatalytic properties of some samples that are mentioned above are not good enough. Therefore, synthesizing MoS$_2$@TiO$_2$ by a simple method and keeping excellent bi-catalytic activity are meaningful.

In this work, P25@MoS$_2$ is synthesized by a simple one-step hydrothermal method. Photodegrading RhB and HER are selected for evaluating the performances of P25@MoS$_2$ in photocatalysis and electrocatalysis, respectively. P25@MoS$_2$ greatly enhances the degrading efficiency due to matched energy levels of P25 and MoS$_2$. Owing to the fact that P25 can hamper the self-assembling of MoS$_2$, more active sites in MoS$_2$ are exposed to improve the HER efficiency. It has been confirmed that both the photocatalytic and electrocatalytic performance of the as-prepared products have been improved, which makes the P25@MoS$_2$ a promising material in the fields of environment and energy.

2. Experimental

2.1. Chemical Reagents

Degussa P25 (80% anatase and 20% rutile) was purchased from Evonik Degussa Company (Shanghai, China). Sodium molybate (NaMoO$_4$·2H$_2$O, AR, 99.0%), hydroxylamine hydrochloride (NH$_2$OH·HCl, AR, 98.5%), and thiourea (CH$_4$N$_2$S, AR, 99.0%) were purchased from Sinopharm
Chemical Reagent Co., Ltd (Shanghai, China). All of the chemicals were used without any further purification, and all of the solutions were prepared with deionized water.

2.2. Synthesis of P25@MoS$_2$

The synthesis process of P25@MoS$_2$ was described, as follows: typically, 90 mg NaMoO$_4$·2H$_2$O, 142 mg CH$_4$N$_2$S, and 52 mg NH$_2$OH·HCl were dissolved in 35 mL deionized water, followed by stirring until clear and homogeneous. The above solution was transferred to a 50 mL Teflon lining-lined stainless steel autoclave. P25 powder was immersed in the solution by stirring and ultrasonic dispersing. Afterwards, the suspension was heated in an electric oven at 200 °C for 24 h. When naturally cooling down to room temperature, the final product was centrifuged and washed several times with deionized water and absolute ethanol, and then dried at 60 °C for 24 h. By adjusting the P25 content, the P25@MoS$_2$ with mass ratios of 2:1, 1:1, and 1:2 were synthesized, and donated as PM21, PM11, and PM12, respectively. Pure MoS$_2$ was synthesized with the same process without P25.

2.3. Characterizations

X-ray diffraction (XRD, Shimadzu7000, Kyoto, Japan) characterization was applied to investigate the crystal phases of the as-synthesized samples. Field emission scanning electron microscopy (FESEM, S-4800, Hitachi, Tokyo, Japan) and high resolution transmission electron microscope (HRTEM, Tecnai G2 F30 S-TWIN, FEI, Hillsboro, OR, USA) were employed to study the morphologies and microscopic structures. X-ray photoelectron spectra (XPS, ESCALAB250Xi, ThermoFisher Scientific, Waltham, MA, USA) were utilized to analyze the chemical components and states. The UV-Vis diffuse reflection spectra (DRS) were recorded by a Shimadzu UV-2600 UV-visible spectrophotometer (Kyoto, Japan). The efficiency of electron–hole pair separation was verified by the photoluminescence spectrophotometer (PL, F-2500, Hitachi, Tokyo, Japan), with an excitation wavelength at 320 nm. The electrochemical workstation (CHI 660E, Chenhua, Shanghai, China) was used to test the activities of hydrogen production.

2.4. Photodegradation Studies

For comparison, the samples of P25, MoS$_2$, PM21, PM11, and PM12 were employed in the photodegradation experiments. 5 mg photocatalysts are suspended in 50 mL RhB solution (15 mg L$^{-1}$), and then the suspends were stirred in the dark for 30 min. to ensure the adsorption-desorption equilibrium. After that, the suspensions were exposed to a 350W Xe lamp (Figure S1) for 120 min. Recycling experiments were carried out while using PM11 for five times under the same condition. Before each recycling, PM11 was washed, centrifuged several times, and then dried for 12 h at 80 °C.

2.5. Hydrogen Production Studies

The samples as the working electrode are prepared, as following: the prepared powder (4.8 mg), acetylene black (1.2 mg), ethanol solution (300 µL), Nafion (5 wt %, 30 µL), and deionized water (300 µL) are added together and ultrasonically treated for 30 min. The electrochemical polarization curves are investigated in a typical three-electrode setup: Ag/AgCl electrode as the reference electrode, the prepared samples on a glassy carbon electrode as the working electrode and the Pt foil (15 mm× 15 mm) as the counter electrode. The experiments are carried out in 0.5M H$_2$SO$_4$ solution.

3. Results and Discussions

3.1. Structure and Morphology Characteristics

Figure 1 shows the XRD patterns of the as-prepared samples. For pure P25, the detected peaks can be assigned to anatase phase TiO$_2$ (JCPDS card No. 21-1272) and rutile phase TiO$_2$ (JCPDS card No. 21-1276). For pure MoS$_2$, all diffraction peaks matched with hexagonal phase MoS$_2$ (JCPDS card No. 37-1492) very well, except for the diffraction peak of (002) plane. The decreased diffraction angle of
(002) plane indicated the increased interplanar distance, which could be ascribed as the ions or particles insertion between layers [24,25]. In Figure 1, with the content of P25 decreases, the intensities of P25 diffraction peaks gradually decreases, implying that the amount of MoS$_2$ increases gradually. Similarly, with the content of MoS$_2$ increases, the diffraction peaks of MoS$_2$ become more and more similar to pure MoS$_2$. Interestingly, the diffraction peak of (002) plane of PM11 (13°) seems more noticeable and closer to standard value (14.4°) than that of PM12 (11°).

Figure 1. X-ray diffraction (XRD) patterns of pure P25, PM21, PM11, PM12, and molybdenum disulfide (MoS$_2$).

To further confirm the structure and morphology of P25@MoS$_2$, FESEM, TEM, and HRTEM characterizations were employed. The FESEM images of pure MoS$_2$ were obtained, as shown in Figure 2a. It is obviously that thin MoS$_2$ nanosheets are self-assembled to form flower-like microspheres. Figure 2b,c depict petal-like structure of PM11, which is apparently different from pure MoS$_2$. This difference should be attributed to the presence of TiO$_2$, which can be further confirmed by the TEM image (Figure 2d) and HRTEM images (Figure 2e,f). Figure 2d presents a typical TEM image of PM11. It evidently displays the compositing of P25 nanoparticles and MoS$_2$ nanosheets. Figure 2e,f are the corresponding HRTEM images for the selected two parts in Figure 2d, respectively. As depicted in Figure 2e,f, the lattice fringes of 0.35 nm can be assigned to the (101) plane of anatase TiO$_2$ [21,26]. Besides, the interlayer spacing of 0.69 nm can be indexed to the (002) plane of 2H-MoS$_2$, which is consistent with the above XRD results (the peak with diffraction angle of 13°) [24]. The TEM and HRTEM results further confirmed the formation of the heterostructure between P25 and MoS$_2$.

The XPS analysis was employed to investigate the chemical composition and bonding configuration of the as-prepared samples. The full XPS survey of PM11 (Figure S2) reveals the existence of Ti, O, Mo, and S, and the observed peak at 285 eV of C 1s originated from the signal of carbon in the instrument [21,23]. Figure 3 displays high-solution spectra. The two peaks located at 465.4 eV and 459.6 eV can be assigned to Ti 2p$_{1/2}$ and Ti 2p$_{3/2}$, respectively, as depicted in Figure 3a. The additional peak located at 473.0 eV, which shifts about 13 eV relative to Ti 2p$_{3/2}$, is considered as a shake-up satellite [27,28]. The O 1s XPS spectrum (shown in Figure 3b) shows four peaks that correspond to four different states of O elements. The peak with the binding energy of 532.4 eV is attributed to adsorbed water (H$_2$O), and the peak with the binding energy of 531.3 eV implies the formation of hydroxyl group (–OH) [29,30]. The peak at around 530.9 eV indicates the formation of Ti–O–Mo bonds between TiO$_2$ and MoS$_2$, and the peak at 530.5 eV is associated with Ti–O bond state of TiO$_2$ [30]. The peak at 226.5 eV in Figure 3c is associated with S 2s, and the other four peaks are attributed to Mo 3d [31,32]. The two strong peaks at 232.7 eV and 229.7 eV correspond to Mo$^{4+}$ 3d$_{3/2}$ and Mo$^{4+}$ 3d$_{5/2}$, respectively [31,32]. The two weak peaks at 236.3 eV and 233.0 eV correspond to Mo$^{6+}$ 3d$_{3/2}$ and Mo$^{6+}$ 3d$_{5/2}$, respectively [31,32]. It has been reported that the appearance of Mo$^{6+}$ implies the complete reduction reaction or oxidization of Mo$^{4+}$ on the surface in air [31,32]. As depicted in Figure 3d, S 2p$_{1/2}$...
and S 2p\(_{3/2}\) are located at 163.7 eV and 162.5 eV, respectively \[21,23\]. The XPS results further confirm that MoS\(_2\) and P25 combine together very well.

![Figure 2](image1.png)

**Figure 2.** (a) Field emission scanning electron microscopy (FESEM) image of pure MoS\(_2\). (b) Low and (c) high magnification FESEM images of PM11 respectively. (d) Transmission electron microscope (TEM) image of PM11. (e,f) High resolution transmission electron microscope (HRTEM) images of PM11 corresponding to the selected areas in (d).

![Figure 3](image2.png)

**Figure 3.** High-resolution X-ray photoelectron spectra (XPS) spectra of PM11: (a) Ti 2p, (b) O 1s, (c) Mo 3d, and (d) S 2p.

3.2. Optical Properties

Figure 4a shows UV-Vis diffuse reflectance spectra. It is obvious that P25 has rather weak absorption in visible range and very high absorption in ultraviolet range. However, MoS\(_2\) and all of the composites (PM21, PM11, PM12) have very high absorption ability in the range of 200–800 nm. Therefore, benefitting from the loading of MoS\(_2\), the utilization of solar spectra for P25 has been broadened.
Figure 4. (a) UV-Vis diffuse reflectance spectra and (b) PL spectra of P25, MoS₂, PM21, PM11, and PM12.

For the semiconductor, the characteristic of charge trapping and recombination of photogenerated electron-hole pairs can be investigated by PL emission spectrum. As exhibited in Figure 4b, the PL curve of P25 has five main emission peaks, which are located at 395.5 nm (3.14 eV), 449.5 nm (2.76 eV), 467 nm (2.66 eV), 481.5 nm (2.58 eV), and 491 nm (2.53 eV), respectively. The peak of 395.5 nm is attributed to electronic transition from the bottom of TiO₂’s conduction band (CB) to the top of TiO₂’s valence band (VB), which implies the recombination of photogenerated electron-hole pairs [21,33]. Additionally, the other four peaks may be associated with the emission of oxygen vacancy formed during the synthesis process [21,33]. As can be seen from Figure 4b, these five peaks that are mentioned above still exist in the samples after loading MoS₂, meaning the presence of P25. However, the intensity of these five peaks is greatly reduced, which implies that the loading of MoS₂ inhibits the recombination of photogenerated electron-hole pairs in P25. The optical performances of the resultant products have confirmed that the compositing between P25 and MoS₂ can improve the absorption of solar light, and suppress the carrier recombination to heighten the separation efficiency of photogenerated electron-hole pairs.

3.3. Photocatalytic Performances

Figure 5 shows the photocatalytic performances for degrading RhB. Without photocatalysts, pure RhB exhibits no appreciable degradation, which indicates the stability of RhB under solar light irradiation. Additionally, pure MoS₂ has a negligible degradation performance. As we know, P25 displays outstanding photodegrading activity, which is usually used as a reference for photocatalytic activity. For the three composites (PM11, PM12, and PM21), they all have higher degradation efficiencies than P25; especially, PM11 has the best photocatalytic performance, which further confirms that the composite of P25 and MoS₂ enhances the photocatalytic performance indeed. In addition, the recycling stability is a very essential ability of photocatalyst for practical application. Figure 5b shows the recycling results of PM11 to degrade RhB. As can be seen in Figure 5b, the PM11 exhibits good stability during five continuous cycles.

Figure 5. (a) The curves of C/C₀ vs. time for photodegrading rhodamine B (RhB). (b) The degradation efficiency in recycling of PM11.
The possible mechanisms (Figure S3) are discussed in the following. When the energy of photon is greater than the band gap (E$_g$) of semiconductor, electron (e$^-$) in the valence band (VB) can be excited to the conduction band (CB) by absorbing photon. Subsequently, hole (h$^+$) is left in VB, and a photo-generated electron-hole pair is produced. The h$^+$ can oxidize H$_2$O to create OH$^•$ species. Meanwhile, e$^-$ can react with O$_2$ (dissolved in solution) to make O$_2$•$^−$ species. Afterwards, the active species for degradation are produced. By loading MoS$_2$ on P25, the absorption range of P25 is broaden, which can be confirmed by the UV-Vis spectra. Therefore, more photon is absorbed to produce more species, hence enhancing the photocatalytic performance of the products. However, electron-hole pairs of TiO$_2$ and MoS$_2$ both have high rates of recombination, which inhibits the production of active species to lower the degrading efficiency [34,35]. Attributed to the matched energy levels, the e$^-$ in CB can be transmitted from MoS$_2$ to P25 and the h$^+$ in VB can be effectively transmitted from P25 to MoS$_2$ [21]. Therefore, the recombination of e$^-$ and h$^+$ is suppressed, and the intensity of the PL spectra is reduced, which is consist with the PL spectra. P25@MoS$_2$ can produce much more active species due to the separation of the electron-hole pairs rapidly. P25@MoS$_2$ is a photocatalyst with promising application prospect, owing to the excellent degradation performance and recycling stability.

3.4. Electro catalytic Performance

The electrocatalytic activity of the prepared samples are studied by HER. Figure 6a shows the comparative polarization curves of those prepared samples. The overpotentials to achieve a current density of 10 mA/cm$^2$ are about 590 mV (P25), 250 mV (MoS$_2$), 275 mV (PM21), 237 mV (PM11), and 215 mV (PM12), respectively. It is clearly shown that P25 needs much higher overpotential than MoS$_2$. This implies that the electrocatalytic performance of P25 is much lower than that of pure MoS$_2$. The overpotential of PM21 is a little higher that of MoS$_2$. PM11 and PM12 have the approximate overpotential and current density changes, which indicates that the electrocatalytic activity of PM11 and PM12 are close.

![Figure 6. (a) Polarization curves and (b) corresponding Tafel plots.](image)

Figure 6b shows the Tafel slope in linear portions are 231.6, 99.6, 113.9, 71.9, and 70.8 mV per decade for P25, MoS$_2$, PM21, PM11, and PM12, respectively. The Tafel slope of P25 is much higher than that of MoS$_2$, confirming that the electrocatalytic performance of pure MoS$_2$ is much higher than that of P25. The Tafel value of PM21 is much lower than that of P25 and higher than that of MoS$_2$. Additionally, PM11 and PM12 hold similar Tafel slopes and are lower than all other simples. All of the results display that MoS$_2$ that is composited with P25 with an appropriate proportion can improve the performance of HER.

It has been proved that the active sites of MoS$_2$ only locate at the edges but not basal plane [20]. For pure MoS$_2$, it grows freely in solution (without P25) to form larger aggregated cluster with less edges. However, in the solution with P25, the MoS$_2$ grows on P25 in the form of scattered nanosheets, improving the amount of edges, thereby the active sites. Therefore, the HER performance is enhanced. In composites, more P25 that is associated with less MoS$_2$ means less aggregated MoS$_2$, which is
Nanomaterials 2019, 9, 1636 advantageous for HER. However, it also means that less MoS$_2$ provides less active sites, which is disadvantageous for HER. More MoS$_2$ is related with MoS$_2$ aggregated seriously, leading to reduced active sites. Hence, the content balance between P25 and MoS$_2$ should be appropriate. Among the three composites, PM21 containing the least MoS$_2$ should be responsible for the lowest HER. However, PM12 with the most content of MoS$_2$ in composites has little HER improvement than PM11, which should be attributed to more aggregated MoS$_2$ in PM12 than in PM11.

4. Conclusions

Through a facile one-step hydrothermal process, P25 loaded few layered MoS$_2$ nanosheets were successfully synthesized. Photocatalysis and electocatalysis performances of the as-prepared products were both investigated. Under simulated sunlight irradiation, P25@MoS$_2$ exhibits enhanced photocatalytic performance. Improved HER performance had displayed that P25@MoS$_2$ had excellent electrocatalytic activity. All of the results have depicted that PM11 possesses excellent bi-catalytic activities. The possible mechanisms of enhanced bi-catalytic activities for PM11 had been discussed. The effective separation and transformation of photogenerated electrons and holes should charge for the promoted performance of photocatalysis. Additionally, the structure of scattered nanosheets of MoS$_2$ with more active sites might be responsible for the enhanced HER performance. The excellent multifunction makes PM11 a promising candidate in environmental protection, as well as in energy conversion, storage, and transport.

Supplementary Materials: The following are available online at http://www.mdpi.com/2079-4991/9/11/1636/s1, Figure S1: The schematic diagram of the photocatalytic system: (1) and (2) cooling water entrance, (3) and (4) cooling water outlet, (5) quartz well, (6) Xe lamp, (7) quartz reactor, (8) magnetic stirrer. The distance between the quartz reactor and lamp is about 12 cm, and the optical density of light source is about 120 mW/cm$^2$, Figure S2: The full scan XPS spectrum of the PM11, Figure S3: Schematic illustration of the mechanisms in degrading process.

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References

1. Nakata, K.; Fujishima, A. TiO$_2$ photocatalysis: Design and applications. J. Photochem. Photobiol. C 2012, 13, 169–189. [CrossRef]

2. Hong, Y.; Tian, C.; Jiang, B.; Wu, A.; Zhang, Q.; Tian, G.; Fu, H. Facile synthesis of sheet-like ZnO assembly composed of small ZnO particles for highly efficient photocatalysis. J. Mater. Chem. A 2013, 1, 5700–5708. [CrossRef]

3. Li, Y.; Feng, X.; Li, Z. Visible-light-initiated Sonogashira coupling reactions over CuO/TiO$_2$ nanocomposites. Catal. Sci. Technol. 2019, 9, 377–383. [CrossRef]

4. Costi, R.; Saunders, A.E.; Elmalem, E.; Salant, A.; Banin, U. Visible Light-Induced Charge Retention and Photocatalysis with Hybrid CdSe–Au Nanodumbbells. Nano Lett. 2008, 8, 637–641. [CrossRef]

5. Zhang, Z.-G.; Liu, H.; Wang, X.-X.; Zhang, J.; Yu, M.; Ramakrishna, S.; Long, Y.-Z. One-step low temperature hydrothermal synthesis of flexible TiO$_2$/PVDF@MoS$_2$ core-shell heterostructured fibers for visible-light-driven photocatalysis and self-cleaning. Nanomaterials 2019, 9, 431. [CrossRef]

6. Sun, Q.; Hu, X.; Zheng, S.; Zhang, J.; Sheng, J. Effect of calcination on structure and photocatalytic property of N-TiO$_2$/g-C$_3$N$_4$@diatomite hybrid photocatalyst for improving reduction of Cr(VI). Environ. Pollut. 2019, 245, 53–62. [CrossRef]
7. Yao, W.; Zhang, B.; Huang, C.; Ma, C.; Song, X.; Xu, Q. Synthesis and characterization of high efficiency and stable Ag3PO4/TiO2 visible light photocatalyst for the degradation of methylene blue and rhodamine B solutions. J. Mater. Chem. 2012, 22, 4050. [CrossRef]

8. Liang, H.; Meng, Q.; Wang, X.; Zhang, H.; Wang, J. Nanoplasmonically engineered interfaces on amorphous TiO2 for highly efficient photocatalysis in hydrogen evolution. ACS Appl. Mater. Inter. 2018, 10, 14145–14152. [CrossRef]

9. Peng, Y.-K.; Keeling, B.; Li, Y.; Zheng, J.; Chen, T.; Chou, H.-L.; Puchtler, T.J.; Taylord, R.A.; Tsang, S.C.E. Unravelling the key role of surface feature behind facetedependent photocatalysis of anatase TiO2. Chem. Commun. 2019, 55, 4415–4418. [CrossRef]

10. Song, T.; Zeng, G.; Zhang, P.; Wang, T.; Ali, A.; Huang, S.; Zeng, H. D reticulated carbon nitride materials high-uniformly capture, black phosphorus as 3D/0D composites for stable and efficient photocatalytic hydrogen evolution. J. Mater. Chem. A 2019, 7, 503–512. [CrossRef]

11. Barbir, F. PEM electrolysis for production of hydrogen from renewable energy sources. Sol. Energy 2005, 78, 661–669. [CrossRef]

12. Gregory, D.P.; Pangborn, J.B. Hydrogen Energy. Annu. Rev. Energy 1976, 1, 279–310. [CrossRef]

13. Popczun, E.J.; McKone, J.R.; Read, C.G.; Biacchi, A.J.; Wiltrout, A.M.; Lewis, N.S.; Schaak, R.E. Nanostructured nickel phosphide as an electrocatalyst for the hydrogen evolution reaction. J. Am. Chem. Soc. 2013, 135, 9267–9270. [CrossRef] [PubMed]

14. Kong, D.; Wang, H.; Lu; Z.; Cui, Y. CoSe2 Nanoparticles grown on carbon fiber paper: An efficient and stable electrocatalyst for hydrogen evolution reaction. J. Am. Chem. Soc. 2014, 136, 4897–4900. [CrossRef]

15. Zheng, Y.; Jiao, Y.; Zhu, Y.; Li, J.H.; Han, Y.; Chen, Y.; Du, A.; Jaroniec, M.; Qiao, S.Z. Hydrogen evolution by a metal-free electrocatalyst. Nat. Commun. 2014, 5, 3783. [CrossRef]

16. Kwon, I.S.; Kwak, I.H.; Abbas, H.G.; Seo, H.W.; Seo, J.; Park, K.; Park, J.; Kang, H.S. Two Dimensional MoS2 Meets Porphyrin via Intercalation to Enhance the Electrocatalytic Activity Toward Hydrogen Evolution. Nanoscale 2019, 11, 3780–3785. [CrossRef]

17. Ohno, T.; Sarukawa, K.; Tokieda, K.; Matsumura, M. Morphology of a TiO2 photocatalyst (Degussa, P-25) consisting of anatase and rutile crystalline phases. J. Catal. 2001, 203, 82–86. [CrossRef]

18. Luo, Y.; Li, X.; Cai, X.; Zou, X.; Kang, F.; Cheng, H.-M.; Liu, B. Two-Dimensional MoS2 Confined Co(OH)2 Electro catalysts for Hydrogen Evolution in Alkaline Electrolytes. ACS Nano 2018, 12, 4565–4573. [CrossRef]

19. Zhang, X.; Liang, Y. Nickel hydr (oxy) oxide nanoparticles on metallic MoS2 nanosheets: A synergistic electrocatalyst for hydrogen evolution reaction. Adv. Sci. 2017, 5, 1700644. [CrossRef]

20. Hinnemann, B.; Moses, P.G.; Bonde, J.; Jørgensen, K.P.; Nielsen, J.H.; Horch, S.; Chorkendorff, I.; Nørskov, J.K. Biomimetic hydrogen evolution: MoS2 nanoparticle as catalyst for hydrogen evolution. J. Am. Chem. Soc. 2005, 127, 5308–5309. [CrossRef]

21. Zheng, L.; Han, S.; Liu, H.; Yu, P.; Fang, X. Hierarchical MoS2 nanosheet@TiO2 nanotube array composites with enhanced photocatalytic and photocurrent performances. Small 2016, 12, 1527–1536. [CrossRef] [PubMed]

22. Song, X.; Chen, G.; Guan, L.; Zhang, H.; Tao, J. Interfacial engineering of MoS2/TiO2 hybrids for enhanced electrocatalytic hydrogen evolution reaction. Appl. Phys. Express 2016, 9, 095801. [CrossRef]

23. Ma, B.; Guan, P.-Y.; Li, Q.-Y.; Zhang, M.; Zang, S.-Q. MOF-derived Flower-like MoS2@TiO2 nanohybrids with enhanced activity for hydrogen evolution. ACS Appl. Mater. Inter. 2016, 8, 26794–26800. [CrossRef]

24. Huang, G.; Chen, T.; Chen, W.; Wang, Z.; Chang, K.; Ma, L.; Huang, F.; Chen, D.; Lee, J.Y. Graphene-like MoS2/Graphene composites: Cationic surfactant-assisted hydrothermal synthesis and electrochemical reversible storage of lithium. Small 2013, 9, 3693–3703. [CrossRef] [PubMed]

25. Acerce, M.; Voiry, D.; Chikourall, M. Metallic 1T phase MoS2 nanosheets as supercapacitor electrode materials. Nat. Nanotech. 2015, 10, 313–318. [CrossRef] [PubMed]

26. Xiang, Q.; Yu, J.; Jaroniec, M. Synergetic effect of MoS2 and graphene as cocatalysts for enhanced photocatalytic H2 production activity of TiO2 nanoparticles. J. Am. Chem. Soc. 2012, 134, 6575–6578. [CrossRef] [PubMed]

27. Werner, A.; Bludovskiy, P.; Selzer, C.; Koch, U.; Giebeler, L.; Oswald, S.; Kaskel, S. Hierarchical Ti-beta obtained by simultaneous desilication and titanation as efficient catalyst for cyclooctene epoxidation. ChemCatChem 2017, 9, 3860–3869. [CrossRef]

28. De Boer, D.K.G.; Haas, C.; Sawatzky, G.A. Exciton satellites in photoelectron spectra. Phys. Rev. B 1984, 29, 4401–4419. [CrossRef]
29. Liu, X.; Xing, Z.; Zhang, H.; Wang, W.; Zhang, Y.; Li, Z.; Wu, X.; Yu, X.; Zhou, W. Fabrication of 3D mesoporous black TiO$_2$/MoS$_2$/TiO$_2$ nanosheets for visible-light-driven photocatalysis. *ChemSusChem* **2016**, *9*, 1118–1124. [CrossRef]

30. Yu, J.; Zhao, X. Effect of surface treatment on the photocatalytic activity and hydrophilic property of the sol-gel derived TiO$_2$ thin films. *Mater. Res. Bull.* **2001**, *36*, 97–107. [CrossRef]

31. Liu, H.; Hu, H.; Wang, J.; Niehoff, P.; He, X.; Paillard, E.; Eder, D.; Winter, M.; Li, J. Hierarchical ternary MoO$_2$/MoS$_2$/heteroatom-doped carbon hybrid materials for high-performance lithium-ion storage. *Chem. Electro. Chem.* **2016**, *3*, 922–932. [CrossRef]

32. Sun, Y.; Hu, X.; Luo, W.; Huang, Y. Self-assembled hierarchical MoO$_2$/graphene nanoarchitectures and their application as a high-performance anode material for lithium-ion batteries. *ACS Nano* **2011**, *5*, 7100–7107. [CrossRef] [PubMed]

33. Zhang, Z.-G.; Liu, H.; Zhang, B.; Zhang, J.; Liu, R.-Z.; Ning, X.; Long, Y.-Z. Synthesis and application of highly ordered arrays of TiO$_2$ rods grown on electrospun PVDF fibers. *Mater. Res. Express* **2017**, *4*, 075907. [CrossRef]

34. Eda, G.; Yamaguchi, H.; Voiry, D.; Fujita, T.; Chen, M.; Chhowalla, M. Photoluminescence from chemically exfoliated MoS$_2$. *Nano Lett.* **2011**, *11*, 5111–5116. [CrossRef]

35. Gao, H.; Wang, H.; Jin, Y.; Lv, J.; Xu, G.; Wang, D.; Zhang, X.; Chen, Z.; Zheng, Z.; Wu, Y. Controllable fabrication of immobilized ternary CdS/Pt-TiO$_2$ heteronanostructures toward high performance visible-light driven photocatalysis. *Phys. Chem. Chem. Phys.* **2015**, *17*, 17755–17761. [CrossRef]

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