ABSTRACT

Objective: To investigate the mechanical properties of different classifications of composites indicated for posterior application as functions of the storage condition and of the energy dose. Material and Methods: Specimens (8x2x2 mm) were obtained according to the factors: I) Composites (3M ESPE): Filtek P60, Filtek Z350XT, and Filtek Silorane; II) Syringe storage conditions: room temperature, aged, oven, refrigerator, and freezer; and III) Energy dose: 24 J/cm² and 48 J/cm². After photoactivation, the specimens were stored at 37°C for 24 h. After storage, a three-point bending test was carried out in a universal testing machine at 0.5 mm/min. Flexural strength (S) and flexural modulus (E) were calculated. Data were analyzed by three-way ANOVA and Tukey’s test (α=0.05). Results: Different storage conditions significantly affected the silorane composite for S; conversely, no effects were noted in terms of E. The accelerated aging protocol significantly increased the S of Filtek P60 and Filtek Silorane, whereas storage in the oven significantly decreased the S for all of the composites tested. Filtek P60 was the only composite not affected by the lower storage temperatures tested for S, whereas for the silorane this parameter was impacted at the same conditions. The factor “dose” was not statistically significant. Conclusions: The syringe storage at different temperature conditions proved to influence mostly the flexural strength, a clinically important characteristic considering the posterior indication of the materials tested. The silorane composite should not be stored at lower temperatures.

Keywords: Composite resins. Physical properties. Storage of substances, products and materials. Aging.

INTRODUCTION

Advances in the mechanical properties of resin composites, associated with improvement in adhesive systems, have favored the clinical application of these restorative materials, especially in posterior teeth. In spite of these improvements, the mechanical properties of these materials vary considerably because of the differences in the composition of the organic matrix as well as the type, size, and packing of the inorganic fillers. The introduction of new or modified dental products on the market requires the assurance that these materials can be stored for an extended period without any decrease in their performance that may affect safety and efficacy. Shelf life is defined as the term or period during which a commodity remains suitable for the intended use. On the other hand, expiration date is the termination of shelf life, after which a percentage of the product, i.e., medical devices, may no longer function as originally intended. In order to determine whether a certain product requires a shelf life and assign an expiration date, there are a number of different parameters that must be considered. In this way, this product must be analyzed to determine if it is susceptible to degradation that would lead to functional failure and to determine the level of risk.
that the failure would present\(^4\).

Accelerated aging protocols are generally applied to new products to provide experimental data in order to estimate their clinical performance and shelf life claims. In these protocols, products are subjected to stresses that are more severe or more frequently applied than normal environmental or operational stresses for a shorter time\(^14\). Studies on accelerated aging combined with basic stability information on the components (functional chemical groups organized in diverse ways - crystalline or amorphous - along with additives such as antioxidants, inorganic fillers, plasticizers, colorants, and processing aids\(^14\) may be used to support tentative expiration dates\(^5\). These variations, combined with the different protocols of clinical applications and oral environment, determine the degradation process\(^14\). In a previous study\(^6\), it was pointed out that the material condition may affect the propagation stage (second stage of the polymerization reaction attributed to the propagating radical), in which the free radicals react successively with monomers to form a three-dimensional network.

Clinicians tend to store the restorative materials at low temperatures, especially the resin composites, in order to “prolong” their shelf life\(^15\). However, these cooled restorative materials are used immediately after being removed from the refrigerator\(^10\). In addition, their manipulative properties are modified, considering that some of the composites are too sticky and tend to flow less at lower temperatures. A consensus that the rheological properties of resin-composites play an important role in terms of restoration longevity exists\(^17\). The viscosity of a resin-composite is directly related to its handling characteristics (placement and shaping on the restored site)\(^19\). In spite of the composite stickiness, manufacturers have modified the composite viscosity, as this manipulative characteristic may allow the formation of macroscopic spaces and microscopic porosities in the restorations\(^23\). This can be partially attributed to the “pull-back” effect\(^1\). Clinically, an unpolymerized resin-composite is applied and exposed while in a varied viscous state and variations in the conversion values can be reached as the rheological properties of composites influence the radical mobility, depending on the viscosity\(^6\). In this way, no objective criteria to support the indication of high viscous composites or “compactable” composites have been provided\(^16\). In addition, the influence of the energy dose as a function of the material condition needs further investigation. As the kinetics of polymerization is dependent upon the energy dose, it could be expected that the longer light exposure time would somehow allow a higher end conversion\(^7\), irrespective of the material condition.

This study evaluated the mechanical properties (flexural strength and flexural modulus) of different classifications of composites indicated for posterior application as functions of the storage condition.

| Composite          | Classification | Organic matrix                              | Fillers                  | Filler type                  | % fillers/wt (vol.)* | Lot # (expiration date) |
|--------------------|----------------|---------------------------------------------|--------------------------|-----------------------------|----------------------|-------------------------|
| Filtek P60 (P60)   | Hybrid         | BisGMA, UDMA, BisEMA                        | Silica /                 | 83.0                        | N126976               |
|                    |                |                                             | (average 0.6 µm)         |                             |                      |                         |
| Filtek Z350XT (XT) | Nanofilled     | BisGMA, TEGDMA, UDMA, BisEMA(6), PEGDMA    | Silica: 20 nm            | Zirconia / Silica           | 72.5                 | 9YE                     |
|                    |                |                                             |                          |                             |                      |                         |
| Filtek Silorane (FS)| Microhybrid   | Silorane**                                  | Quartz,                  | 76.0                        | N128528               |
|                    |                |                                             |                          |                             |                      |                         |

BisGMA: bisphenol-glycidyl-methacrylate; UDMA: urethane-dimethacrylate; BisEMA: bisphenol-a-ethoxy dimethacrylate; TEGDMA: triethyleneeglycol dimethacrylate; PEGDMA: Poly(ethylene glycol) dimethacrylate
*Manufacturer’s information (3M ESPE, St. Paul, MN, USA)
**Bis-3,4-Epoxy cyclohexylethylPhenyl-Methylsilane and 3,4-Epoxy cyclohexylcyclopoly methylsiloxane
Note: The brand name Filtek Silorane is used in other countries as Filtek LS and Filtek P90

Figure 1- Description of the composites used in this study

References:
1. Mechanical properties of composites as functions of the syringe storage temperature and energy dose.
2. Assessed aging protocols are generally applied to new products to provide experimental data in order to estimate their clinical performance and shelf life claims.
3. Studies on accelerated aging combined with basic stability information on the components (functional chemical groups organized in diverse ways - crystalline or amorphous - along with additives such as antioxidants, inorganic fillers, plasticizers, colorants, and processing aids) may be used to support tentative expiration dates.
4. Clinicians tend to store the restorative materials at low temperatures, especially the resin composites, in order to “prolong” their shelf life.
5. However, these cooled restorative materials are used immediately after being removed from the refrigerator.
6. In addition, their manipulative properties are modified, considering that some of the composites are too sticky and tend to flow less at lower temperatures.
7. A consensus that the rheological properties of resin-composites play an important role in terms of restoration longevity exists. The viscosity of a resin-composite is directly related to its handling characteristics.
8. In spite of the composite stickiness, manufacturers have modified the composite viscosity, as this manipulative characteristic may allow the formation of macroscopic spaces and microscopic porosities in the restorations.
9. This can be partially attributed to the “pull-back” effect. Clinically, an unpolymerized resin-composite is applied and exposed while in a varied viscous state.
10. Variations in the conversion values can be reached as the rheological properties of composites influence the radical mobility, depending on the viscosity.
11. In this way, no objective criteria to support the indication of high viscous composites or “compactable” composites have been provided.
12. In addition, the influence of the energy dose as a function of the material condition needs further investigation.
13. As the kinetics of polymerization is dependent upon the energy dose, it could be expected that the longer light exposure time would somehow allow a higher end conversion, irrespective of the material condition.
14. This study evaluated the mechanical properties (flexural strength and flexural modulus) of different classifications of composites indicated for posterior application as functions of the storage condition.
(room temperature, aged, oven, refrigerator, and freezer) and the energy dose. The following research hypotheses were tested: (1) the storage temperature can influence the flexural strength of the composites tested; (2) the flexural modulus of the composites stored at different temperatures will be inferior to that observed at room temperature; (3) the energy dose influences the mechanical properties of the composites, irrespective of the storage temperature.

MATERIAL AND METHODS

Experimental design

In this in vitro study, mechanical characterizations were performed according to the factors: (1) composites, at three levels: Filtek P60 (P60), Filtek Z350XT (XT), and Filtek Silorane (FS); and (2) material conditions, at five levels: I – control (room temperature - RT, 22°C), II – accelerated aged composites (AA), III – oven at 37°C for 24 h (OV), IV – refrigerator at 2°C for 24 h (RF), and V – freezer at -15°C for 24 h (FZ); (3) energy dose, at two levels: I – 24 J/cm², II – 48 J/cm². The characteristics of the resin composites selected are described in the Figure 1. The schematic representation of the experimental design is depicted in Figure 2. Thirty experimental groups were obtained of the product among the combinations of the factors under study.

Accelerated aging protocol

The composites underwent a simplified protocol for accelerated aging (also called the "10-degree rule") according to the collision theory based on the Arrhenius model. The composite syringes were stored at 37°C in an oven for 12 weeks. According to Clark5 (1991), the accelerated aging protocol can be calculated by means of a mathematical formula, as follows:

\[
r = Q_{10} ^{(RT - ET)/10}
\]

where,

- \( r \) = accelerated aging rate;
- \( RT \) = room temperature (22°C);
- \( ET \) = elevated temperature (37°C);
- \( Q_{10} \) = reaction rate coefficient (2).

According to this protocol, for each 10°C increase above the room temperature, the reaction rate doubles5. In this way, as the \( RT \) was equivalent to 22°C, \( r \) corresponds to \( 2^{1.5} \) (37-22=15; 15/10=1.5). Thus, the accelerated aging rate was based on the increase of 15°C in the storage temperature, which is 1.5 squared (\( r=1.5^2=2.83 \)). Finally, in order to estimate the aging time equivalent to that which would occur at room temperature, \( r \) was multiplied

![Figure 2- Schematic representation of the experimental design](image-url)
by the storage time (12x2.83), which represents approximately 9 months of aging. It is important to mention that the same lot numbers were used for the material condition levels of "room temperature" and "aged". In addition, care was taken to ensure that the composites were not expired after the accelerated aging protocol (simulating a 9-month shelf life)⁷.

**Specimen fabrication**

After the storage time respective to the experimental groups had elapsed, the syringes were kept untouched until room temperature was reached. An infrared thermometer (General Tools & Instruments, IRT-206 Gun-style Infrared Thermometer, New York, NY, USA) was used to evaluate the syringe temperatures. Flexural strength evaluations differed slightly from that described in ISO 4049. The composites were applied to a Teflon mold (8×2×2 mm), positioned over a polyester strip (n=10). After filling the mold to excess, the material surface was covered with a Mylar strip and a glass slide and compressed to extrude excess material. All restorative materials were photoactivated using an LED light-curing unit (Bluephase, Ivoclar Vivadent, Schaan, Liechtenstein) with a power density of 1200 mW/cm² for 20 s (energy dose of 24 J/cm²) or for 40 s (energy dose of 48 J/cm²). Prior to testing and throughout the experiment, the power density was monitored using a handheld radiometer (Model 100, Demetron Research Corp., Danbury, CT, USA). All of the composites selected were shade A3 (an exception was XT: shade A3B). Specimens were then stored in distilled water at 37°C for 24 h.

**Flexural strength test (S)**

Prior to the test, the specimen dimensions were measured using a digital caliper (Digimatic Caliper CD, Mitutoyo, Japan). The three-point bending test was carried out in a universal testing machine (Instron model 3342, Instron Corp., Canton, MA, USA) at 0.5 mm/min and 5 mm span between supports. S (s) was calculated as follows:

\[
s = \frac{3F \times L}{2b \times h^2}
\]

Flexural Modulus (E) was calculated by:

\[
E = \frac{L^3}{4b \times h^2} \times \frac{F}{Y}
\]

where F is the maximum strength in N; L the distance between the rests; b the width of the specimen; h the height of the specimen; and F/Y the slope of the linear part of the stress–strain curve. Statistical analysis of S and E was performed with three-way ANOVA and Tukey post-hoc test (α = 5 %).

**RESULTS**

Figure 3 shows data of S regarding the three factors: composite x syringe storage condition x energy dose. The results of S are listed in Table 1. The highest S mean was noted for P60 AA (329.2 MPa), and the lowest for Z250 OV (117.3 MPa). The interaction between the factors “composites” and “material condition” was significant. However, the factor “dose” and the triple interaction among the factors were not significant. All of the composites exhibited statistically
equivalent $S$ means at room temperature (RT) and freezer (FZ) storage ($p>0.05$). The $S$ means of P60 were not influenced by the storage at lower temperatures (RF and FZ), whereas for the silorane composite $S$ mean significantly decreased ($p<0.05$). On the other hand, only the storage at FZ significantly decreased $S$ means of XT; the $S$ mean was not influenced by the RF storage in comparison to that at RT ($p>0.05$). All the composites exhibited significantly decreased in the $S$ means when the syringes were stored in the oven at 37°C for 24 h compared to that of at RT ($p<0.05$). On the other hand, the simplified protocol of accelerated aging led to a significant increase in the $S$ means for P60 and for Filtek Silorane ($p<0.05$).

Figure 4 shows the data of $E$ regarding the three

![Flexural Modulus Graph](image)

**Figure 4** - Graphic of flexural modulus (GPa) considering all factors: I. Composites: Filtek P60 (P60), Filtek Silorane (FS) and Filtek Z350XT (XT); II. Syringe storage conditions: Room Temperature (RT), Accelerated aging (AA), Oven 24 h (OV), Refrigerator 24 h (RF), Freezer 24 h (FZ); III. Energy doses: 24 J/cm$^2$ and 48 J/cm$^2$

### Table 1 - Mean of flexural strength ($S$) regarding interaction Composite x Syringe storage condition

| Storage condition          | Filtek P60 | Filtek Silorane | Filtek Z350XT |
|----------------------------|------------|-----------------|---------------|
| Room Temperature (RT)      | 224.9 (41.2)$^{abA}$ | 227.1 (42.7)$^{aA}$ | 208.8 (27.7)$^{aA}$ |
| Accelerated aging (AA)     | 329.2 (43.0)$^{abA}$ | 290.6 (59.7)$^{abB}$ | 255.3 (52.4)$^{AB}$ |
| Oven 24 h (OV)             | 143.2 (24.6)$^{cA}$ | 117.3 (24.1)$^{cB}$ | 156.1 (21.4)$^{cA}$ |
| Refrigerator 24 h (RF)     | 231.2 (51.0)$^{abA}$ | 169.1 (27.4)$^{cB}$ | 202.1 (37.5)$^{cA}$ |
| Freezer 24 h (FZ)          | 191.7 (35.7)$^{cA}$ | 167.9 (15.7)$^{cA}$ | 163.4 (30.7)$^{cA}$ |

Mean values followed by different capital letters in row and small letters in column: significant ($p<0.05$)

( ) : standard deviation

### Table 2 - Mean of flexural modulus ($E$) regarding interaction Composite x Syringe storage condition

| Storage condition          | Filtek P60 | Filtek Silorane | Filtek Z350XT |
|----------------------------|------------|-----------------|---------------|
| Room Temperature (RT)      | 16.9 (2.5)$^{abA}$ | 15.4 (2.3)$^{aA}$ | 16.9 (1.2)$^{abA}$ |
| Accelerated aging (AA)     | 17.6 (2.2)$^{abA}$ | 14.2 (1.5)$^{aA}$ | 19.0 (3.6)$^{abA}$ |
| Oven 24 h (OV)             | 14.6 (4.1)$^{cA}$ | 14.9 (4.2)$^{cA}$ | 13.3 (2.4)$^{cA}$ |
| Refrigerator 24 h (RF)     | 19.3 (1.9)$^{abA}$ | 15.9 (1.9)$^{cA}$ | 19.2 (1.4)$^{abA}$ |
| Freezer 24 h (FZ)          | 21.5 (1.2)$^{cA}$ | 18.3 (1.7)$^{cA}$ | 21.0 (1.9)$^{cA}$ |

Mean values followed by different capital letters in row and small letters in column: significant ($p<0.05$)

( ) : standard deviation
Mechanical properties of composites as functions of the syringe storage temperature and energy dose

The interaction between the factors “composites” and “material condition” was significant. The factor “dose” and the triple interaction among the factors were not significant. The factor “storage condition” produced no effect in the $E$ means of the silorane composite ($p>0.05$). For P60 and XT, storage at OV significantly decreased the $E$ means, whereas the storage at FZ significantly increased the flexural modulus ($p<0.05$).

DISCUSSION

Although the composites tested presented equivalent $S$ means at room temperature, the results were material-dependent when considering the different storage conditions. Filtek Silorane was the composite most influenced by the storage condition. The simplified protocol of accelerated aging produced an increase in $S$ for the silorane composite whereas all of the other conditions (oven, refrigerator, and freezer for 24 h) produced a significant reduction in this mechanical parameter. On the other hand, Filtek P60 was not influenced by either of the two lower temperatures whereas Filtek Z350XT showed a significant reduction in $S$ only when stored at the freezer at 37°C for 24 h. Aging the Filtek P60 produced a significant increase in $S$. The composite Filtek Z350XT was the only composite not influenced by the simplified protocol of accelerated aging. Storing all of the composites in the oven produced a significant reduction in $S$. Thus, the first research hypothesis raised in the present study was upheld by the experimental data as the storage temperature influences the flexural strength of the composites tested.

The simplified approach for accelerated aging used in the present study was based on conducting testing at a single accelerated temperature and then employing the rule stating that the rate of a chemical reaction will increase by a factor $Q_{10}$ (usually of 2), which represents a doubling of the reaction rate for every 10°C increase in temperature. This type of conservative relationship is indicated for a variety of medical polymers that have been previously characterized. The degradation may be primarily chemical – a combination of effects arising from oxidative chain scission, oxidation hydrolysis, changes in crystallinity, and other factors that may be environmentally dependent. Chemical crosslinking links molecules, while crystallinity and fillers introduce physical limitations to monomer mobility. Under these variations, the rate of radical propagation eventually becomes diffusion-limited and the polymerization rate decelerates, often providing only a limited conversion even in the presence of an unreacted monomer and a population of radicals. In this way, the crosslinking will be restricted by the molecular response, depending on the storage temperature, since the degree of freedom for monomer motions is reduced as the these monomers become irreversibly linked. This was particularly true in the accelerated aging composites as significant variations in the flexural strength occurred in comparison to that at room temperature.

In contrast to those results noted after the accelerated aging protocol, in which the $S$ means increased, the $S$ means obtained after storage in the oven at 37°C for 24 h significantly decreased for all of the tested composites. One would argue that an enhanced flexural strength would be obtained after the storage in the oven at 37°C as the result of an increased conversion given by temperature-derived mobility of polymeric chains and consequent entrapped radical engagement. Therefore, it is reasonable to infer that the composites underwent a chemical degradation at shorter storage time (24 h) before a change in the crystallinity would occur throughout the 12-week storage at the same temperature (aging protocol). It could be argued that, by significantly increasing the flexural strength after aging the composite syringes of Filtek P60 and Filtek Silorane would provide better clinical performance. Conversely, the increased flexural strength may not represent improved mechanical properties, despite the equivalence in terms of flexural modulus when compared to that of the means at room temperature. In this way, it was clear that these contradictory results when comparing shorter and longer storages in the oven at 37°C have in common the fact that both storage conditions interfere in the composite stability.

Many aspects must be considered when analyzing the results. It is important to highlight that the mechanical properties of composites tested depend on the properties of their constituent materials. Thus, when comparing the different composites, the large ranges in the $S$ means were due to the different classifications in the composite formulations, although all tested materials are indicated for restoring posterior teeth. Filtek Silorane is filled with a combination of fine quartz particles and radiopaque yttrium fluoride, being classified as a microhybrid resin composite. The concentration of filler particles in this composite is 76.0% by weight. The filler in Filtek P60 is zirconia/silica in a concentration of 83.0% by weight, classified as a hybrid composite. Filtek Z350XT is a nanofilled composite with 72.5% of fillers. As these materials are essentially polymeric in nature, their laboratorial and clinical performances are related to the rate of degradation of their inherent properties.
structure and configuration over time. In contrast to methacrylate-base composites, the silorane-based composite contains cationic ring-opening monomers, which represent a compensating mechanism for shrinkage stress achieved during polymerization. This monomer system was obtained from the reaction of oxirane and siloxane molecules. This resin composite combines two main advantages: low polymerization shrinkage, due to the ring-opening oxirane, and an increased hydrophobicity due to the siloxane compound. The silorane-based composite revealed a decreased water sorption, solubility, and associated diffusion coefficient compared to those observed when methacrylate-based composites were tested.

The elastic modulus of low-shrinking composites and methacrylate-based composite was recently reported as being similar, despite differences in the formulation, even though the silorane composite has higher molecular weight and higher filler content. In another study, it was also revealed that the silorane material presents relatively higher flexural strength/modulus and fracture toughness but relatively lower compressive strength and hardness than some methacrylate-based restorative materials. In the present study, no significance was noted among the composites in terms of $E$ at room temperature. In addition, none of the composites had the means of flexural modulus affected by the accelerated aging protocol. Different from that observed for flexural strength, all of the storage conditions produced no effect on $E$ of the silorane composite. On the other hand, Filtek P60 and Filtek Z350XT were affected by storage in the oven at 37°C for 24 h, significantly decreasing and Filtek Z350XT was affected by storage in the freezer significantly increased $E$ ($p<0.05$). In this way, the second research hypothesis, that the energy dose influences in the mechanical properties of the composites, irrespective of the storage temperature, was rejected. According to the statistical analysis, the factor “dose” and the triple interaction among the factors were not significant for both the parameters tested. A reasonable explanation was that, in the present study, a 24 J/cm² energy dose was sufficient to guarantee the composite polymerization of 2-mm thick specimens. Clinically, it would be important to obtain well polymerized composites with enhanced mechanical properties. On the other hand, an energy dose of 48 J/cm² should be used with caution by clinicians considering the increase in the polymerization stress generated when higher energy doses are applied. In addition, the specimens exposed to a low energy dose were stored at 37°C for 24 h after photoactivation, and this fact may allow an increased monomer conversion and enhanced mechanical properties similar to that of obtained when a higher energy dose was applied.
previous study\textsuperscript{7}, it was pointed out that the energy
dose associated with the bonding approach are
determinant factors in terms of the presence of
internal gaps for both the methacrylate-based
composite and for the silorane-based one.

The present study deals with a clinical relevant
issue as the syringe storage conditions tested
influenced the mechanical properties of different
classifications of resin composites. Storage as a
function of the temperature proved to influence
mostly the composite flexural strength, possibly
due to changes in the chemical stability and/or
crystallinity (especially at higher temperatures)
of the composites and also due to the impact of
the storage temperature and the transportation
conditions, their effect on the composite clinical
performance can be minimized if properly considered
by the clinicians\textsuperscript{8}. Extrapolations to clinically support
performance can be minimized if properly considered
conditions, their effect on the composite clinical
the storage temperature and the transportation
temperatures). Although the dental manufacturers
may not be able to control all of the variables like
the storage temperature and the transportation
conditions, their effect on the composite clinical
performance can be minimized if properly considered
by the clinicians\textsuperscript{8}. Extrapolations to clinically support
and validate the results using different resin-
composite categories and filler contents need to be
done with caution since future studies are required to
evaluate the performance of these composites.

**CONCLUSION**

Within the limitations of this study, the following
can be concluded:

The storage condition influences the flexural
strength and the flexural modulus of the composites
in comparison to that observed at room temperature.

The energy dose does not influence the
mechanical properties, irrespective of the storage
condition.

The aging protocol improves the flexural
strength of Filtek P60 and Filtek Silorane, whereas
storage in the oven at 37°C for 24 h impacts the
same parameter for all of the composites tested.

It seems that the silorane composite should
not be stored at lower temperatures, whereas the
Filtek P60 is not impacted by these temperatures.

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