Non-covalent interactions of Cysteine onto C60, C59Si, and C59Ge: A DFT study

Mohsen Doust Mohammadi
Tehran University: University of Tehran

hewa Y abdullah (✉ hewayaseen@gmail.com)
Tishk International University  https://orcid.org/0000-0001-5766-3858

Research Article

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Non-covalent interactions of Cysteine onto C\textsubscript{60}, C\textsubscript{59}Si, and C\textsubscript{59}Ge: A DFT study

Mohsen Doust Mohammadi\textsuperscript{a}, Hewa Y. Abdullah\textsuperscript{b,\ast}

\textsuperscript{a} School of Chemistry, College of Science, University of Tehran, Tehran 14176, Iran
\textsuperscript{b} Physics Education Department, Faculty of Education, Tishk International University, Erbil, 44001, Iraq

\ast Corresponding author Email: hewayaseen@gmail.com

Abstract

The study of intermolecular interactions is of great importance. This study attempted to quantitatively examine the interactions between Cysteine (C\textsubscript{3}H\textsubscript{7}NO\textsubscript{2}S) and fullerene nanocages, C\textsubscript{60}, in a vacuum. As the frequent introduction of elements as impurities into the structure of nanomaterials can increase the intensity of intermolecular interactions, nanocages doped with silicon and germanium have also been studied as adsorbents C\textsubscript{59}Si and C\textsubscript{59}Ge. Quantum mechanical studies of such systems are possible in the density functional theory (DFT) framework. For this purpose, various functionals, such as B3LYP-D3, \omega\textsubscript{B}97XD, and M062X, have been used. One of the most suitable basis functionals for the systems studied in this research is 6-311G (d), which has been used in both optimization calculations and calculations related to wave function analyses. The main part of this work is the study of various analyses that reveal the nature of the intermolecular interactions between the two components introduced above. The results of conceptual DFT, natural bond orbital, non-covalent interactions, and quantum theory of atoms in molecules were consistent and favored physical adsorption in all systems. Germanium had more adsorption energy than other dopants. The HOMO–LUMO energy gaps were as follows: C\textsubscript{60}: 5.996, C\textsubscript{59}Si: 5.309, and C\textsubscript{59}Ge: 5.188 eV at B3LYP-D3/6-311G (d) model chemistry. The adsorption sensitivity increased when an amino acid molecule interacted with doped C\textsubscript{60}, and this capability could be used to design a nanocarrier to detect Cysteine amino acids.

Keyword: Boron Nitride; Cysteine; C\textsubscript{3}H\textsubscript{7}NO\textsubscript{2}S; Density Functional Theory; Wave function analysis.
Introduction

The first attempts to come up with the concept of intermolecular interactions were made by Clausius [1], van der Waals [2], and London [3,4]. Intermolecular interactions can be divided into long-range and short-range classes. The long-range class includes electrostatic, induction, and dispersion forces, and it varies with the inverse powers of the distance $r^{-n}$, which is the reciprocal of the intermolecular distance. Conversely, the short-range class includes exchange and repulsion forces that decrease exponentially with distance, as in $e^{-\alpha r}$. Both repulsive and attractive electrostatic forces arise from classical Coulombic interactions between the charge distributions of two molecules. These forces are also pairwise additive and anisotropic. The interacting molecules cause instantaneous fluctuations in the electron distribution, and such a disruption creates dispersion forces that are pairwise additive and always attractive. Induction forces, which are non-additive and attractive, are created by the distortion of the distribution of molecular charges resulting from the influence of the electric fields of other molecules. Exchange and repulsion are both non-additive and of opposite signs [5].

Ab initio calculations are a reliable method for obtaining polarizabilities (i.e., a set of constants to show the charge redistribution when a molecule is exposed to an electric field) and electric multipole moments used to explain the long-range forces [6]. In the case of short-range forces, the theory is rather more complicated [7] due to the overlap of electron densities between molecules. The previous long-range theory should be modified in terms of short-range penetration [8-10], charge transfer [11], and damping effects [12-14]. Generally, full quantum methods can only be well implemented in small systems, but there are technical problems with large molecules [15]. The development of wave function analysis methods has helped predict the nature of intermolecular interactions, even in the case of large systems. Although the ab initio
methods are few, neat, and classifiable, there are countless wave function analysis methods that have not yet been systematically categorized in the scientific literature.

After Feynman’s famous “There’s Plenty of Room at the Bottom” speech in 1959 in Caltech, the history of nanoscience has seen a dramatic expansion. Early works on carbon-based nanomaterials by Iijima [16,17] and Novoselov [18-21] paved the way for the development of nanotechnology in the use of inorganic nanomaterials. Among these, the structures of boron nitride are of special importance due to their extraordinary physical and chemical properties [22-24]. Studies on the optoelectrical, mechanical, and thermal properties [25] of such structures are still under consideration [26,27]. Furthermore, these nanomaterials are widely used, for example, as nanocarrier [28-39] or nanocarriers of drugs [40-42]. These processes occur through adsorption (adhesion of a particle to a surface), and they are primarily related to the intermolecular interactions between the adsorbent and the adsorbate.

This study examined the adsorption of Cysteine amino acid molecule onto the surface of pristine, silicon-, and germanium-doped fullerene nanocages, $C_{60}$, $C_{59}Si$, and $C_{59}Ge$, respectively. All structures, such as isolated species and complex systems, were optimized using various functionals, including B3LYP-D3, ωB97XD, M062X, and 6-311G(d) basis set. The most stable structure in terms of electronic energy was chosen for further wave function analyses using the B3LYP-D3/6-311G(d) model chemistry. We referred to the conceptual density functional theory (DFT), natural bond orbital (NBO), non-covalent interactions (NCI), and quantum theory of atoms in molecules (QTAIM) from among the most important and reliable wave function analyses ever developed. In this study, they were used to understand the nature of intermolecular interactions. All studies were performed in vacuum; therefore, the sensitivity and reactivity of the adsorbent were appreciated. The main ideas of each analysis were explained in
the shortest possible way, as these materials are scattered in the available resources, and it is a little difficult for beginners to have them together.

Computational details

The Kohn–Sham DFT framework was implemented in this study. Each structure, including amino acid and nanocages, was geometrically optimized in a vacuum using the so-called Berny geometry optimization algorithm developed by H. B. Schlegel in 1982 [43]. The PBE0 functional [44,45], which is the hybrid–exchange correlation form of the Perdew–Burke–Ernzerhof generalized gradient approximation (GGA) [46], meta-hybrid GGA functional M06-2X developed by the Truhlar group [47,48], the Head–Gordon group functional ωB97XD [49], which includes dispersion and long-range corrections, and the Becke 3-parameter Lee–Yang–Parr functional with Grimme dispersion correction B3LYP-D3 [50-52] were employed. The MC-311 or 6-311G (split-valence triple-zeta) basis set in the Gaussian package [53] has been developed by many individuals [54-62]. In the current work, the 6-311(d) basis set was used (i.e., d-type Cartesian–Gaussian polarization functions). Split-valence basis sets allow orbitals to alter the size but not the shape. When polarized functions are added to the basis sets, this limitation is removed by including orbitals with an angular momentum greater than what is needed for the ground state to depict each atom. This basis set is large enough and well able to simulate molecular orbitals. Data from benchmark studies also confirm this fact [63-67]. GaussView 6.0.16 [68] and ChemCraft [69] packages were used for building molecules. Linux-based Gaussian 16 software Rev. C.01 [53] was used for self-consistent field (SCF) calculations (as link 502). The default convergence criteria remained intact during the calculations. The SCF convergence was considered by comparing the maximum force and maximum displacement with the threshold values (0.00045 Hartree/Bohr and 0.0018 Bohr). Moreover, no symmetry
limitations were imposed on the optimization process. Wave function stability, frequency checks, and zero-point energy corrections (ZPECs) were considered to ensure the accuracy of the calculations. Stability calculations guarantee that this optimized electronic wave function may be minimal in the wave function space rather than a saddle point and that it is completely different from finding minima or saddle points on a nuclear potential energy surface. As the Gaussian software includes the NBO version 3.1 software [70-72] (as link 607), it is used to perform population analysis studies. The Multiwfn [73] package developed by Tian Lu is fed by formatted Gaussian checkpoint files for various wave function analyses. In this work, Multiwfn was used for NBO, NCI, and QTAIM studies. O’Boyle et al. developed the cclib and GaussSum [74] package to obtain the DOS diagrams.

To calculate the adsorption energy ($E_{\text{ads}}$) of the two molecules (nanocage and CYSTEINE), the following underlying relation is applied:

$$E_{\text{ads}} = E_{\text{cage/amino}} - E_{\text{cage}} - E_{\text{amino}} + \Delta E_{(\text{BSSE})} + \Delta E_{(\text{ZPE})}$$  \hspace{1cm} (1)

where $E_{\text{cage/amino}}$ is the energy of the cluster, and $E_{\text{amino}}$ and $E_{\text{cage}}$ are the energies of the isolated amino acid and nanocage, respectively. The negative values of $E_{\text{ads}}$ (i.e., exothermic adsorptions) show that the formed amino acid/nanocage cluster is stable. The electron density of each nucleus can be determined using a function centred on another nucleus. Therefore, in all structures, the quality of the basis set is not the same. This means that the basis set of one molecule can be effective in compensating for the violation of the basis set of another molecule. This effect is called the basis set superposition error (BSSE). There are two well-known methods for the BSSE correction, (1) chemical Hamiltonian approach (CHA) [75] and Boys and Bernardi’s counterpoise (CP) correction procedure [76,77,15] which CP method is employed in this work:

$$\Delta E_{(\text{BSSE})} = \Delta E_{\text{cluster}} - \Delta E_{\text{cage}} - \Delta E_{\text{amino}}$$  \hspace{1cm} (2)
The ZPECs are calculated using Eq. 3:

\[
\Delta E_{\text{ZPE}} = \Delta E_{\text{cluster (ZPE)}} - \Delta E_{\text{cage (ZPE)}} - \Delta E_{\text{amino (ZPE)}}
\]

(3)

Result and discussion

3.1 Methodology

The DFT concept was initiated by Thomas [78], Fermi [79], and Dirac [80] in 1927–1930 using uniform electron amino acid energy densities. In 1951, Slater [81] opened a new horizon, which Hohenberg, Kohn, and Sham [82,83] turned into a complete theory in 1964–1965. In DFT, the overall energy is shown in terms of the full electron density instead of the complicated \( N \) electron wave function. DFT is considered an exact theory (not a model like the Hartree–Fock model), but the approximations that are applied have caused it to fail; nevertheless, it is an efficient and usable tool [84]. The Hartree–Fock model and Kohn–Sham DFT are presented as follows [85,84]:

\[
E_{\text{HF}} = V + \langle h\rho \rangle + \frac{1}{2} \langle \rho J(\rho) \rangle - \frac{1}{2} \langle \rho K(\rho) \rangle
\]

(4)

\[
E_{\text{KS}} = V + \langle h\rho \rangle + \frac{1}{2} \langle \rho J(\rho) \rangle + E_{\text{X}}[\rho] + E_{\text{C}}[\rho]
\]

(5)

where \( V \) is the nuclear repulsion energy, \( \rho \) is the density matrix, \( \langle h\rho \rangle \) is the one-electron (kinetic plus potential) energy, \( \frac{1}{2} \langle \rho J(\rho) \rangle \) is the classical electron repulsion, \( -\frac{1}{2} \langle \rho K(\rho) \rangle \) is the exchange energy, \( E_{\text{X}}[\rho] \) is the exchange functional, and \( E_{\text{C}}[\rho] \) is the correlation functional. By comparing relations 4 and 5, in Kohn–Sham formalism, the Hartree–Fock model is essentially a special case of DFT, where \( E_{\text{X}}[\rho] = -\frac{1}{2} \langle \rho K(\rho) \rangle \) and \( E_{\text{C}}[\rho] = 0 \). Accordingly, in DFT, the methods differ in having different functionals \( E_{\text{X}} \) and \( E_{\text{C}} \).
Unfortunately, exact functionals for exchange and correlation are not known; therefore, approximations have historically been developed to approach the real values of physical quantities, such as local-density approximation (LDA), generalized gradient approximations (GGA) \cite{86,46}, meta-GGA, and hyper-GGA. In this respect, a number of functionals include only the exchange part, such as Becke 88 (B88) \cite{86} and Gill96 (G96) \cite{87}, and some of them include a pure correlation part, such as Vosko–Wilk–Nusair (VWN5) \cite{88}, Perdew 86 (P86) \cite{89}, Lee–Yang–Parr (LYP) \cite{90}, Perdew–Zunger (PZ81) \cite{91}, Cole–Perdew (CP) \cite{92}, and Perdew–Wang (PW92) \cite{93}. Conversely, Becke 97 (B97) \cite{94}, Perdew–Burke–Ernzerhof (PBE-PBEPBE) \cite{44}, and Perdew–Wang 91 (PW91PW91) \cite{95} are examples of exchange–correlation functionals (see Head-Gordon et al. article for a general overview of the types of functionals to a climb of Jacob’s ladder \cite{64}). The origin of the differences among the $E_{xc}[\rho]$ arise from exchange-correlation holes (i.e. an exclusion area around electron or a “no-fly zone” which prevents other electron to penetrate, each functional has its own XC fingerprint).

In the present study, the hybrid functionals PBE0 and B3LYP-D3 were used. Note that the D3 in B3LYP-D3 refers to the D3 version of Grimme’s dispersion with Becke–Johnson damping (GD3BJ) \cite{52}. In hybrid functions, the exchange energy is divided into two parts to improve performance. The first part is the result of accurate calculations obtained from Hartree–Fock calculations, and the next part of the exchange energy is calculated using the DFT method. B3LYP uses three mixing parameters:

\[
E_{xc} = A \times E_x(\text{LSDA}) + (1-A) \times E_x(\text{HF}) + B \times \Delta E_x(\text{B88}) \\
+ E_{c}(\text{VWN}) + C \times E_{c}(\text{LYP}) \\
\text{(6)}
\]

and PBE0 consists of a 1:3 mixture of Hartree-Fock (HF) and DFT exchange energies:

\[
E_{xc} = 0.25 \times E_x(\text{HF}) + 0.75 \times E_x(\text{PBE}) + E_c(\text{PW91}) \\
\text{(7)}
\]
B3LYP has a weakness in considering charge transfer excitations, as it is not a long-range corrected functional [96]. This means that it uses a constant of 20% HF exchange for both short and long ranges. CAM-B3LYP [97] and ωB97XD are range-separated hybrid functionals [98] that are suitable for excitation energy calculations and conjugated systems [99]. ωB97XD is a range-separated version of the B97 functional with Grimme’s D2 dispersion model (addition of an empirical C_6·R^6 dispersion term). In short-range interactions, it comprises 22% HF exchange contributions and 100% HF in the long-range. A standard error function is implemented to describe the intermediate region using a range separation parameter (i.e., ω = 0.2–0.3 Bohr^{-1}) [100]. The range-separated term consists of two parts which splits the DFT exchange interaction as follows:

\[
\frac{1}{r_{12}} = \frac{\alpha + \beta \text{erfc}(\omega r_{12})}{r_{12}} + \frac{\alpha + \beta \text{erf}(\omega r_{12})}{r_{12}}; \quad \text{erfc} \equiv 1 - \text{erf}
\]  

(8)

The first term in Eq. 8 shows short-range contribution which is singular and decays to zero on a length scale of \(\sim 1/\omega\) and the second term is related to non-singular long-range part. The \(\alpha\) and \(\beta\) parameters allow HF-exchange and DFT exchange incorporate over whole range. M06-2X, a meta-hybrid GGA functional with double non-local exchange (2X) amounts, belongs to the M06 suite of functionals with 54% HF contributions, in which the non-interacting kinetic energy density is used as input to the functional and the electron density and its gradient [101]. This functional is well defined for systems with dispersion forces.

**Geometric surveys**

In the first step, molecular geometries were constructed using GussView software. The method of making nanocages is explained fully in the Help section of the software. Creating a Cysteine amino acid molecule does not take much effort, but for more certainty, geometry
related to amino acid molecules was prepared using the PubChem online database [102]. At the beginning of the research, we chose models that were faster and less expensive. All structures, including isolated molecules and clusters, were initially examined using PBE0/6-311G(d) model. As our ultimate goal was to examine intermolecular forces, we used models that calculated intermolecular interactions more accurately. The optimization process was repeated with the same basis set and functionals M06-2X, B3LYP-D3, and ωB97XD. The chemistry of the different models varied within the trade-offs made between computational cost and accuracy.

Once geometric optimization was completed, the presence or absence of imaginary frequencies (i.e., frequencies that have negative values do not represent the minimum and actually represent a transition state) was examined. Thus, in each optimized structure, frequency calculations were performed. In the output file of the Gaussian software, the keyword Nimag was assigned for this purpose. A value of zero (Nimag = 0) means that there are no negative frequencies according to the number of negative eigenvalues of the Hessian matrix. Frequency calculations were performed to determine the ZPEC, which must be added to the total energy.

To study the interactions between nanocages and a amino acid molecule, a C\textsubscript{60} cage with 1.45 Å bond length was selected. The optimized C\textsubscript{60} structure was used to prepare a doped nanocage. For this purpose, doped nanocages were obtained by substituting silicon and germanium instead of carbon atom and they were re-optimized in the same optimization process.

Each nanocage is illustrated in Fig 1, which shows the position of the doped elements compared with the pure nanocage. The injection of the doped elements slightly changed the length of the bonds, indicating that the electronic structure was altered. These changes in the electronic structure led to differences in properties. Doping is a critical and successful strategy for detecting the properties of a nanomaterial as an adsorbent [103]. The dopants alter the
sensing properties by changing the HOMO-LUMO gap and morphology, creating more centres
for amino acid interaction on the adsorbent surface.

As shown in Fig 2, there are five sites on the nanocage that have the potential to absorb the
amino acid molecule. The top of the carbon atom is represented by $T_1$. The position of $T_2$ is
related to the placement of the amino acid molecule on top of the bond between two hexagonal
rings. The $T_3$ position is the placement of the amino acid molecule on top of the bond between
hexagonal and pentagonal rings. The $T_4$ position is located above the hexagonal ring. And $T_5$ is
on top of the pentagonal ring. Structurally, the Cysteine molecule has four heads, each of which
can be placed on any of the five adsorption sites in C60 (Fig 3).

The starting point of the study is placing the amino acid molecule on the absorbent surface
in different positions with different angles and distances and with different orientations and
optimizing the whole cluster system. Finding the global minimum is the most challenging step of
the study. Although we may never get it right away in the computing process, various strategies
can be used to find many local minima and choose the most stable mode based on their
quantitative value. If we have the full potential energy surface (PES) of two fragments, we can
easily report the global minimum. However, having the PES requires thousands of calculations,
which are practically impossible. The approach followed in this research is based on placing the
amino acid molecule on the adsorbent surface at different distances and orientations. For this
purpose, a low-cost method, such as PM6, was used. Dozens of initial orientations were selected
to achieve the most stable structure in terms of energy content. The PEB0/6-311G(d) model
chemistry was used to optimize the obtained stable structure and repeat the process using the
abovementioned functionals.
Fig 1. The values of bond length for (a) C$_{60}$, (b) C$_{59}$Si, and (c) C$_{59}$Ge. The optimization process has been done using the B3LYP-D3/6-311G (d) level of theory.
The relaxed cluster structures obtained from B3LYP-D3/6-311G (d) are illustrated in Fig 4. When the value of the energy of adsorption is below the range of chemical interest (i.e., from the third decimal place onwards following the decimal point), the results are identical [104]. Table 1 lists the values obtained from the calculations of the four different methods for optimizing complex structures. The quantitative amounts of adsorption energies obtained from the ωB97XD and B3LYP-D3 models were similar. Moreover, PBE0 data showed how different the data would be if the dispersion effect was not considered. Among the four different models, the values obtained from the B3LYP-D3/6-311G (d) model showed the highest stability, based on the absolute value of the highest absorption energy. As indicated in Table 1, among the doped elements, germanium produced the highest absorption energy (-1.121 eV).

To understand the nature of intermolecular forces, whether they are strong or weak or are the result of van der Waals or electrostatic effects, the wave function is analyzed in the following sections. According to Table 1 and based on our previous experience [105,106], the B3LYP-D3 functional is a better match to the systems studied in this paper. Thus, it was used for the wave function analysis calculations.

### Table 1

The interaction energy ($E_{\text{ads}}$) for C$_{60}$, C$_{59}$Si, and C$_{59}$Ge with Cysteine molecule. All values are in (eV).

| Systems            | PBE0  | B3LYP-D3 | M06-2X | ωB97XD |
|--------------------|-------|----------|--------|--------|
| Cysteine/C$_{60}$  | -0.291| -0.555   | -0.488 | -0.520 |
| Cysteine/C$_{59}$Si| -0.469| -0.629   | -0.604 | -0.677 |
| Cysteine/C$_{59}$Ge| -0.773| -1.121   | -0.957 | -1.019 |
Fig 2. The $(T_x)$ positions including on top of carbon atom $(T_1)$, on top of the bond between two hexagonal rings $(T_2)$, on top of the bond between hexagonal and pentagonal rings $(T_3)$, on top of the hexagonal ring $(T_4)$, and top of the pentagonal ring $(T_5)$.

Fig 3. The values of bond length for Cysteine molecule. The optimization process has been done using the B3LYP-D3/6-311G (d) level of theory.
Fig 4. The most stable form of (a) Cysteine/C$_{60}$, (b) Cysteine/C$_{59}$Si, and (c) Cysteine/C$_{59}$Ge. All clusters have been optimized using the B3LYP-D3 functional and 6-311G(d) basis set.
Conceptual DFT descriptors

The conceptual DFT descriptors, including chemical potential (μ), chemical hardness (η), electronegativity (χ), electrophilicity (ω), and nucleophilicity (1/ω), are implemented to consider the chemical reactivity/stability of certain atomic sites of molecules, the charge transfer through nucleophilic or electrophilic attacks, and the analysis of chemical reactions [107-109]. According to the basic principles of DFT, energy is a functional of the wave function (a functional can be used to turn a function into a number [110]), and the changes in the number of particles, ΔN, and external potential, Δν(ρ), can describe a typical process. Therefore, if we have the ground state energy, $E[N, ν(ρ)]$, of an N electron system, the energy of the perturbed system can be obtained as follows [111]:

$$\Delta E = E[N + ΔN, ν(ρ) + Δν(ρ)] - E[N, ν(ρ)]$$

$$= \left( \frac{∂E}{∂N} \right)_ν ΔN + \int \left( \frac{∂E}{∂ν(ρ)} \right)_N δν(ρ)d(ρ) + \frac{1}{21} \left( \frac{∂^2 E}{∂N^2} \right)_ν ΔN^2 + 2 \int \left( \frac{∂}{∂N} \left( \frac{∂E}{∂ν(ρ)} \right)_N \right) ΔN δν(ρ)d(ρ) + \int \left[ \int \left( \frac{∂^2 E}{∂ν^2(ρ)} \right)_N δν(ρ)δν(ρ')d(ρ)drdr' \right] + ...$$

The above Taylor expansion is usually truncated in the second order. The coefficients of the expansion have specific meanings in chemical language. For example, for non-degenerate states, perturbation theory describes electron density, $ρ(ρ)$, as follows [112,113]:

$$\left( \frac{∂E}{∂ν(ρ)} \right)_N = ρ(ρ)$$

Parr et al. [114] proved that μ is a quantitative measure of the tendency of an electron to pull out a system. The direction is from high to low μ values. The electronic chemical potential (μ) and electronegativity (χ) relation are expressed as Eq. 10

$$-χ = \left( \frac{∂E}{∂N} \right)_{ν(ρ)} = μ ± \left( ε_{LUMO} + ε_{HOMO} \right) \frac{1}{2} (IP + EA)$$

Pearson [115] showed hardness (η) as a measure of resistance of a system to a change in the electronic cloud as follows:
Similarly, the Fukui function (which shows the change of \( \rho(r) \) at a given position by changing of the number of electron, \( N \)) \cite{116,117}, \( f(r) \), and the dual descriptor \cite{118}, \( \Delta f(r) \) are defined as follows:

\[
f(r) = \left( \frac{\partial^2 E}{\delta \nu(r) \partial N} \right)_{r(r)} = \left( \frac{\partial \mu}{\delta \nu(r)} \right)_{N} \tag{13}
\]

\[
\Delta f(r) = \left( \frac{\partial^3 E}{\delta \nu(r) \partial^2 N} \right)_{r(r)} = \left( \frac{\partial \eta}{\delta \nu(r)} \right)_{N} \tag{14}
\]

Electrophilicity (\( \omega \)) \cite{119} is not the coefficient of the above-mentioned expansion. At the equilibrium point (\( \mu = 0 \)), \( \omega \) is the stabilization energy gained by a system (i.e., \( \omega \) is the capacity to accept an arbitrary number of electrons). In addition, a maximum number of electrons gained by the system \( \Delta N_{\text{max}} \) \cite{120} prove that a typical system can be stabilized by gaining electrons when \( \mu < 0 \) and \( \eta > 0 \). Electrophilicity (\( \omega \)) and \( \Delta N_{\text{max}} \) are defined as

\[
\omega = \frac{\mu^2}{2\eta} \approx \frac{(IP + EA)^2}{4(IP - EA)} \tag{15}
\]

\[
\Delta N_{\text{max}} = -\frac{\mu}{\eta} \tag{16}
\]

As shown in the above equations, the HOMO–LUMO energy gap (HLG) \cite{121} is related to ionization potential (\( IP \)) and electron affinity (\( EA \)). Theoretically, based on the Koopmans’ \cite{122} and Janak’s \cite{123} approximations, the ionization potential is equal to the negative value of HOMO, \( (e_{\text{HOMO}} = -IP) \), and the electron affinity is equal to the negative value of LUMO, \( (e_{\text{LUMO}} = -EA) \). It should be noted that HOMO and LUMO can only be obtained from HF or DFT (i.e. single determinant methods) and the LUMO has no contribution to the total energy of
the system. In addition, calculating the LUMO values is basis set sensitive; however, methods have been developed that can eliminate the LUMO dependence on the basis set [124].

The values of these descriptors are presented in Table 2. The energy gap ($E_g$) of C60 was 5.996 eV using the B3LYP-D3/6-311G (d) model chemistry, and the adsorption of Cysteine on it reduced the energy gap to 5.067 eV. The silicon- and germanium-doped nanocages also reduced the HLG values. Molecules with large HLG had high values of $\eta$, whereas those with low HLG had low $\eta$. Table 2 shows that the LUMO values became more negative after the adsorption of Cysteine onto the nanocages; therefore, HLG was reduced. Reactivity of species was obvious due to the values of the descriptors $\eta$ and $\mu$. Conductivity arose from a high level of reactivity, and the charge transfer between the cages and adsorbate could be implemented through an chemical device as a carrier. To obtain a visual image of how the energy levels, especially HOMO and LUMO, are positioned, the total density of state maps is used in Fig 5.

**Table 2**

HOMO energy ($\mathcal{E}_H$), LUMO energy ($\mathcal{E}_L$), HOMO–LUMO energy gap (HLG), chemical potential ($\mu$), chemical hardness ($\eta$), and electrophilicity ($\omega$). All values are in eV and were obtained using the B3LYP-D3/6-311G (d) level of theory.

| Systems       | $\mathcal{E}_H$ | $\mathcal{E}_L$ | HLG  | $\mu$  | $\eta$ | $\omega$ |
|---------------|-----------------|-----------------|------|--------|--------|----------|
| C$_{60}$      | -8.176          | -2.179          | 5.996| -5.177 | 2.998  | 40.183   |
| C$_{59}$Si    | -7.848          | -2.540          | 5.309| -5.194 | 2.654  | 35.805   |
| C$_{59}$Ge    | -7.954          | -2.766          | 5.188| -5.360 | 2.594  | 37.266   |
| Cysteine/C$_{60}$ | -6.705        | -1.639          | 5.067| -4.172 | 2.533  | 22.045   |
| Cysteine/C$_{59}$Si | -7.065    | -1.783          | 5.282| -4.424 | 2.641  | 25.839   |
| Cysteine/C$_{59}$Ge | -7.487    | -2.428          | 5.059| -4.957 | 2.529  | 31.080   |
Fig 5. Density of state maps for (a) C$_{60}$ and Cysteine/C$_{60}$, (b) C$_{59}$Si and Cysteine/C$_{59}$Si, and (c) C$_{59}$Ge and Cysteine/C$_{59}$Ge. Data were obtained from the B3LYP-D3/6-311G (d) model chemistry.
**NBO analysis**

Canonical molecular orbitals (MO) lead to delocalized descriptions of electrons derived directly from SCF methods [125]. Delocalization means that the MO obtained from several electrons belongs to adjacent atoms. As this is physically difficult to imagine, to create physical intuition, we had to use a method that would give us an understanding. As the wave function is invariant under unitary transformations, localization is possible. The NBO [70] method, developed by Weinhold et al., is an intuitive method used to reach localized and understandable orbitals (i.e., localized Lewis-like electronic structures) using the first-order reduced density matrix of the wave function.

Various types of NBO analyses offer tremendous power to display bond order, Lewis structures, donor–acceptor interactions, charge transfer, resonance weights, bond type, and hybridization. The validity of NBO analysis extends from atoms to supermolecules, spanning the entire periodic table. The term “natural” was introduced by Löwdin [126]. NBO analysis uses natural orbitals, as NBOs are obtained as local block eigenfunctions of the density matrix. “Bond orbital” means that the NBO algorithm calculates the distribution of electron density in bonds between atoms. Generally, NBO methods serve as a bridge between wave function and elementary valency and bonding concepts.

Providing a complete orthogonal set of natural atomic orbitals (NAOs), localized one-centre orbitals from the atom-centred basis set are the most fundamental parts of the NBO algorithm. To perform a natural population analysis, natural hybrid orbitals (NHOs), NBOs, and natural localized molecular orbitals (NLMOs) must be obtained from NAOs to transform a given wave function into localized orbitals that correspond to the one-centre (“lone pair”) and two-centre (“bond”) elements of the Lewis structure picture:
For a (σ) bond located between atoms A and B, the NBO is defined as follows:

$$\sigma_{AB} = C_A h_A + C_B h_B$$ \hspace{1cm} (17)

where $C_A$ and $C_B$ are the corresponding polarization coefficients (for covalent bonds $C_A = C_B$; if the electronegative A is assumed to be greater than B, we have: $C_A >> C_B$). The $h_A$ and $h_B$ are natural hybrid valance orbitals. To complete the span of valance space, an antibonding NBO ($\sigma^*$) is also defined as follows:

$$\sigma^*_{AB} = C_A h_A - C_B h_B$$ \hspace{1cm} (18)

In this study, NBO methods were utilized to measure the bond order. Coulson’s definition of chemical bonding has paved the way for many chemical bond analyses [127]. The most popular methods used to find the bond order are the Mulliken bond order analysis [128] (Equation 19), the Mayer bond order [129-131] (Equation 20), and the Wiberg bond index (WBI) in Löwdin orthogonalized basis [132,133] (Equation 21). Therefore,

$$I_{AB} = \sum_i \sum_{a\in A; b\in B} 2C_{a,b}^i C_{b,a}^i S_{a,b} = 2 \sum_{a\in A; b\in B} P_{a,b} S_{a,b}$$ \hspace{1cm} (19)

$$I_{AB} = I^\alpha_{AB} + I^\beta_{AB} = 2 \sum_{a\in A; b\in B} \left[ (P^\alpha S)_{ba} \left( (P^\alpha S)_{ab} + (P^\beta S)_{ba} \right) \right]$$ \hspace{1cm} (20)

$$I_{AB} = \sum_{a\in A; b\in B} P_{ab}^2$$ \hspace{1cm} (21)

where $P$ is the density matrix, and $S$ is the overlap matrix. Compared with Mulliken and Mayer’s bond orders, WBI has less basis set dependence, especially the basis set that includes diffuse functions, and it provides more accurate results. The WBI values are reported in Table 3. Accordingly, we can conclude that $C_{59}$Si and $C_{59}$Ge adsorbents are more active materials in this
study in adsorbing Cysteine compared with pristine C\textsubscript{60}. The WBI values show that the interaction of the amino acid molecule with C\textsubscript{60} can be classified as a weak interaction. Conversely, the interactions between silicon- and germanium-doped C\textsubscript{60} with the Cysteine molecule are stronger than the van der Waals interactions. The results of the WBI are in good agreement with the adsorption energies reported in Table 1.

**Table 3**
The Mayer, Mulliken, and Wiberg bond index obtained for atomic bonds and intermolecular interactions between Cysteine molecule and C\textsubscript{60}, C\textsubscript{59}Si, and C\textsubscript{59}Ge. All calculations were performed using the B3LYP-D3/6-311G(d) level of theory.

| Systems       | Mulliken | Mayer | Wiberg |
|---------------|----------|-------|--------|
| Cysteine/C\textsubscript{60} | 0.039    | 0.088 | 0.201  |
| Cysteine/C\textsubscript{59}Si | 0.232    | 0.298 | 0.450  |
| Cysteine/C\textsubscript{59}Ge | 0.280    | 0.355 | 0.404  |

**QTAIM analysis**

The idea that a molecule is a collection of atoms linked by a network of bonds is not directly related to quantum mechanics and is beyond theoretical definition. Bader’s topological quantum theory of atoms in molecules (QTAIM) analysis [134,135] was developed to answer the question, “What is an atom in a molecule and how does one predict its properties?” Based on the topology of the electron density, the structure of a molecule is revealed by the stationary points of the electron density and the gradient paths that originate and terminate at these points. In the AIM method, electrons are distributed in space in the gravitational field of the nucleus. This means that the nucleus exists as a point in a cloud of negative charge with an electron density distribution. Electron density $\rho(\mathbf{r})$ is a measurable property that determines the shape and properties of atoms and materials in general.
In this theory, a critical point is a minimum or a maximum of electron density ($\nabla \rho(\mathbf{r}) = 0$). A chemical bond is defined as a line between two atoms, in which the electron density is the maximum along with. There are trajectories that both originate and terminate at critical points and are related to each other by a saddle point. The nature of a critical point is defined by a Hessian matrix:

$$H_{ij} = \frac{\partial^2 \rho(r)}{\partial X_i \partial X_j}$$  \hspace{1cm} (22)$$

$$\nabla^2 \rho(r) = \frac{\partial^2 \rho}{\partial x_i^2} + \frac{\partial^2 \rho}{\partial y_i^2} + \frac{\partial^2 \rho}{\partial z_i^2}$$  \hspace{1cm} (23)$$

$$S = \sum_{i=1}^{3} \text{Sign}(\lambda_i)$$  \hspace{1cm} (24)$$

where $\lambda_i$ is the eigenvalue of the Hessian matrix. Critical points are identified by two characteristics: rank ($\lambda_i > 0$ or $\lambda_i < 0$) and signature ($\lambda_i \neq 0$). Critical points with a rank less than 3 are topologically unstable; thus, they become a number of critical points with a rank of 3. Using (R, S) coordinates, there are four types of critical points: atomic critical point (3,−3), bond critical point (BCP) (3, −1), ring critical point (RCP) (3,1), and cage critical point (CCP) (3,3) [136]. The following (Poincaré-Hopf) relation should be satisfied between critical points:

$$n - b + r - c = 1$$  \hspace{1cm} (25)$$

where $n$, $b$, $r$, and $c$ denote the number of nuclei, BCP, RCP, and CCP, respectively. The Laplacian of electron density ($\nabla^2 \rho(\mathbf{r}) = \lambda_1 + \lambda_2 + \lambda_3$) and that of electron density $\rho(\mathbf{r})$ at any BCP provide useful information about the different types of chemical interactions. The amount of $\rho(\mathbf{r})$ in BCP determines the bond strength and the bond order. In the case of the covalent bond, the Laplacian is less than zero ($\nabla^2 \rho(\mathbf{r}) < 0$), and the values of $\rho(\mathbf{r})$ are large, indicating that the charges are concentrated between two nuclei. The positive values of the Laplacian ($\nabla^2 \rho(\mathbf{r}) > 0$)
and the low values of $\rho(r)$ indicate that the charge dissipates in the distance between the two nuclei and that the interactions can be classified as a closed-shell type, which is related to ionic bonds, hydrogen bonds, and van der Waals bonds [137]. In particular, in the case of hydrogen bonding, if the electron density is in the range of 0.035–0.002 and if the Laplacian of electron density is in the range of 0.0139–0.002, the bond can be a hydrogen type.

The values of the Lagrangian kinetic energy $G(r)$, potential energy density $V(r)$, and energy density $H(r) = G(r) + V(r)$ can also be helpful in identifying the type of interactions. For a covalent bond, $H(r) < 0$; $H(r)/\rho(r) >> 0$; and $G(r)/|V|(r) < 0.5$. For non-covalent interactions, $H(r) > 0$; $H(r)/\rho(r) > 0$; and $G(r)/|V|(r) > 1$. The virial theorem [138] suggests a relationship between $G(r)$, $V(r)$, and $\nabla^2 \rho(r)$.

$$\frac{1}{4} \nabla^2 \rho(r) = 2G(r) + V(r)$$  \hspace{1cm} (26)

Another descriptor is the bond elliptical index ($\epsilon$), which is a good criterion for detecting conjugation and hyperconjugation. When $\epsilon$ is large, we have an elliptical structure that indicates that the $\pi$ character is large. If $\epsilon = 0$, the bond is cylindrical (double or triple) and very stable. Here, $\epsilon$ can be used to refer to the stability of the interactions and is defined as follows [139]:

$$\epsilon = \frac{\lambda_1}{\lambda_2} - 1 ; \quad \lambda_1 < \lambda_2 < 0; \text{ and } \lambda_3 > 0$$  \hspace{1cm} (27)
Table 4

The AIM topological parameters, including electron density ($\rho(r)$), Laplacian of electron density ($\nabla^2 \rho(r)$), the kinetic electron density $G(r)$, potential electron density $V(r)$, eigenvalues of Hessian matrix ($\lambda$), and bond ellipticity index ($\varepsilon$) at BCPs of the Cysteine molecule and C$_{60}$, C$_{59}$Si, and C$_{59}$Ge. All values have been calculated using the B3LYP-D3/6-311G(d) level of theory from NBO analysis.

| Systems       | $\rho$ | $\nabla^2 \rho$ | $G(r)$ | $V(r)$ | $G(r)/V(r)$ | $\lambda_1$ | $\lambda_2$ | $\lambda_3$ | $\varepsilon$ |
|---------------|--------|----------------|--------|--------|-------------|-------------|-------------|-------------|-------------|
| Cysteine/C$_{60}$ | 0.0332 | 0.1420         | 0.0348 | -0.0341 | 1.0213      | -0.0258     | -0.0109     | 0.1787      | 1.3792      |
| Cysteine/C$_{59}$Si | 0.0542 | 0.0611         | 0.0375 | -0.0597 | 0.6278      | -0.0467     | -0.0382     | 0.1460      | 0.2233      |
| Cysteine/C$_{59}$Ge | 0.0977 | 0.2660         | 0.1077 | -0.1489 | 0.7234      | -0.1120     | -0.1156     | 0.4936      | 0.0316      |

Considering the values shown in Table 4, the Laplacian electron density $\nabla^2 \rho(r)$ is positive for all interactions between the Cysteine amino acid molecule and all nanocages. Therefore, the interaction between amino acid and nanocages is detected as non-covalent. Fig 6 shows the BCP position. The presence of critical points between the amino acid molecule and the nanocages emphasizes the strong interactions between the two components. Among the nanocages, C$_{59}$Si and C$_{59}$Ge are more capable of adsorbing the amino acid molecule, as the values of $G(r)/|V(r)|$ for both of them are between 0.5 and 1. This means that the interaction tends to be strong in van der Waals. The other descriptors, $H(r)$ and $H(r)/\rho(r)$, agree with these results. The values obtained from $\varepsilon$ show stable intermolecular interactions.
Fig 6. AIM molecular graphs for (a) Cysteine/C₆₀, (b) Cysteine/C₅₉Si, and (c) Cysteine/C₅₉Ge systems. Orange dots represent the boundary critical points (BCPs).
NCI analysis

The results obtained from the previous sections show that the interactions between the amino acid molecule and pure and doped nanocages are non-covalent. Therefore, to verify this, these interactions were examined in terms of NCI analysis to determine the accuracy of the results. NCI analysis uses two functions: (1) \( \text{sign} \lambda_2(r) \rho(r) \) (product of electron density and the sign of the second Hessian eigenvector) and (2) reduced density gradient (RDG), which is a dimensionless form of a gradient of electron density [140]. The values of these two functions are displayed on a two-dimensional coordinate plane: the \( \text{sign} \lambda_2(r) \rho(r) \) is placed on the x-axis, and the RDGs are displayed on the vertical axis. The following equation is used to obtain the RDG values [140,141]:

\[
\text{RDG} = \frac{1}{2(3\pi^2)^{1/3}} \left\{ \frac{\Delta \rho(r)}{\rho(r)^{1/3}} \right\}
\]  

(28)

Depending on the position of the second function in the diagram, three areas are created, indicating the type of interactions. In the \( \text{sign} \lambda_2(r) \rho(r) < 0 \) region, strong non-covalent interactions are found; in the \( \text{sign} \lambda_2(r) \rho(r) \approx 0 \) region, relatively weak van der Waals interactions are defined; in the \( \text{sign} \lambda_2(r) \rho(r) > 0 \) region, repulsion forces are dominant [140,141]. Fig 7 compares the NCI plots for both isolated nanocages and amino acid/nanocages clusters. Repetitive results from previous analyses are also shown in the NCI analysis. That is, in the case of C_{50}Si and C_{59}Ge adsorbents, the adsorption intensity is stronger than in others. To reference this, \( \text{sign} \lambda_2(r) \rho(r) \approx 0 \) and \( \text{RDG} \approx 0.5 \) should be considered.
Fig 7. Plots for the reduced density gradient (RDG) vs. sign($\lambda_2$)$\rho(r)$ values of (a) $C_{60}$ and Cysteine/$C_{60}$, (b) $C_{59}$Si and Cysteine/$C_{59}$Si, and (c) $C_{59}$Ge and Cysteine/$C_{59}$Ge. The data were obtained from B3LYP-D3 /6-311G (d) level of theory. The left side diagrams are isolated nanotubes and the right side diagrams are BCF/nanosheet clusters.
Conclusion

Various wave function analyses were performed to study the intermolecular interactions between the Cysteine amino acid and fullerene nanoadsorbents. The standard model chemistry in this study was B3LYP-D3/6-311G (d), which was implemented for the geometry optimization process and NBO calculations. The PBE0, ωB97XD, and M06-2X functionals and the 6-311G(d) basis set were also used for geometry optimization. Considering the dispersion in these functionals was a factor that completely changed the results of these calculations. In the optimization process, the spatial orientations of the two monomers relative to each other were the determining factors that led to the finding of the local minima. Spanning the entire potential energy surface for such interactions could not be attained. Therefore, it is important to use an appropriate algorithm to find the local minima. Comparing the adsorption energy of different clusters showed that the adsorbents C_{59}Si and C_{59}Ge trapped the amino acid molecule more intensely. The QTAIM and NCI analysis results identified the intermolecular interactions of the type of strong van der Waals interaction for these nanocages. As the amino acid and the mentioned adsorbents interacted well, these nanomaterials could be used to design a nanocarrier for the Cysteine molecule.

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Conflicts of Interest

The authors declare no conflict of interest
*Declarations

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Hewa Y. Abdullah: Conceptualization, Writing - review & editing, Resources, Supervision.)

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