Planar superconductor/ferromagnet hybrids: Anisotropy of resistivity induced by magnetic templates

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We investigated experimentally the transport properties of a superconducting cross-shaped aluminium microbridge fabricated on top of ferromagnetic BaFe12O19 single crystal. It was demonstrated that a one-dimensional domain structure in the ferromagnetic substrate can induce the formation of superconducting channels above magnetic domains. This leads to a giant anisotropy of resistivity of the superconducting bridge, caused by the appearance of continuous paths of supercurrents flowing along domain walls.

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Hybrid superconductor-ferromagnet (S/F) structures have attracted considerable attention in connection with the possibility to control thermodynamic and transport properties of the S/F hybrids by manipulating the magnetic state of the ferromagnetic constituents (1, 2, 3, 4 and references therein). Provided an insulating layer prevents the diffusion of Cooper-pairs from the superconductor to the ferromagnet, the exchange interaction between superconducting and ferromagnetic parts can be effectively suppressed and the interaction between both subsystems occurs via slowly decaying stray magnetic fields. Nonuniform magnetic field, induced by the ferromagnet, can modify the conditions for the appearance of superconductivity due to the effect of a local field compensation, resulting in the field-induced superconductivity (5) and an exotic dependence of the superconducting critical temperature $T_c$ on an applied magnetic field $H$ (6, 7, 8).

An increase of the width of the equilibrium magnetization loop $M(H)$ of the S/F hybrids, compared with plain superconducting films, can be interpreted as an enhanced “magnetic” pinning of vortices by various magnetic textures: periodic arrays of magnetic dots (9, 10) or irregular magnetic bubble domains (11). The magnetostatic interaction between the vortices and the “built-in” magnetic field is also known to lead to the unusual field dependence of the electrical resistance $R(H)$ of the S/F hybrids at temperatures close to the superconducting critical temperature $T_c$ (2, 3, 12, 13, 14).

Recently, electrical transport in S/F hybrids at low temperatures was studied for the following planar structures: Nb/Co (15), Al/CoPd (16), NbSe2/Pt (17), MoGe/Pt (18, 19), Pb/Pt (20). Superposition of the bias current and the supercurrent that is induced by hard ferromagnets may lead to a remarkable change of the current ($I$) – voltage ($V$) characteristics of superconducting bridges (15, 16), what can be interpreted as a “current” compensation effect (21). The tunable alignment of magnetic domains in low-coercive ferromagnetic films, using an in-plane oriented external field $H$, makes it possible to introduce a guided vortex motion in a desirable direction along the domain walls (17, 18, 19, 20).

In this Letter we are aiming at the investigation of the anisotropy of the electrical transport properties of the S/F hybrids, induced by a single straight domain wall. We measured both the magnetoresistance $R(H)$ and the $I – V$ dependencies of a superconducting bridge in two perpendicular directions (i.e. along and across a domain wall in the ferromagnetic substrate).

In order to study the anisotropy of the low-frequency transport properties in a planar S/F hybrid, we prepared a bi-layered sample consisting of a bulk ferromagnetic substrate and a thin-film Al microbridge on top. The ferromagnetic and superconducting parts were electrically connected via a single straight domain wall. The planar S/F hybrid system under investigation is shown in Fig. 1. The top layer shows an atomic force microscopy image (AFM) of the cross-shaped Al bridge (lighter shades). The areas labelled I–IV were used as contact pads for transport measurements. The bottom layer shows a magnetic force microscopy image (MFM) of the ferromagnetic BaFe12O19 substrate. Light and dark regions correspond to the magnetic domains with $M_z > 0$ and $M_z < 0$. Note that the MFM image is vertically extended to illustrate the magnetic domains. Black solid lines depict the edges of the Al bridge.
isolated by a 5 nm SiO2 buffer layer, so that the interaction between these parts can be expected to be exclusively electromagnetic. When cut along the proper crystallographic direction, a ferromagnetic BaFe12O19 single crystal exhibits a one-dimensional (1D) stripe-type domain structure with dominant in-plane magnetization and relatively small out-of-plane component $M_z$ (the bottom image in Fig. 1). Measurements with a vibrating sample magnetometer (VSM) revealed that at low temperatures the magnetization of the used crystal depends almost linearly on the applied perpendicular magnetic field with the slope $dM/dH \approx 3.2 \cdot 10^5$ (A/m)T$^{-1}$ and that it saturates at $H \approx 1.7$ T. This means that external magnetic fields $|H| \leq 80$ mT can only be of minor influence on the domain structure, since the variations of magnetic moment of the substrate are expected to be about 4.5% from the saturated magnetization ($5.5 \cdot 10^5$ A/m).

The location of the domain walls and their shape were determined by magnetic force microscopy (MFM), prior to the preparation of the Al bridge. The expected amplitude of the $z$-component of the nonuniform magnetic field, $B_0$, exceeds the upper critical field $H_{c2}$ of such Al films even at low temperatures (see below). The cross-shaped Al microbridge (50 nm thick) was fabricated by means of e-beam lithography, molecular beam epitaxy and lift-off etching (the top image in Fig. 1). The width $w$ of the ”arms” of the microbridge was equal to 30 $\mu$m and limited by the width of the magnetic domains. Four contact pads, labelled in Fig. 1 as I–IV, were used for the injection of the dc bias current $I$ and for the measurement of the voltage drop $V$ for two different cases: along the domain wall ($V_{||}$) using the electrodes I and II and across the domain walls ($V_{\perp}$) using the electrodes III and IV. This symmetrical form of the superconducting element was intentionally chosen in order to have the possibility to compare the $I$–$V$ characteristics in two perpendicular directions for the same magnetic landscape.

Figure 2 shows the level curves of the dc resistance of the sample, $V_{||}(H,T)/I_0 = 0.8 R_n$ and $V_{\perp}(H,T)/I_0 = 0.8 R_n$, $R_n$ being the normal-state resistance, $I_0 = 100$ $\mu$A. These lines can be commonly interpreted as the dependence of the superconducting critical temperature $T_c$ on $H$. In spite of some inessential differences, both phase transition lines $T_c^0 (H)$ and $T_{c\perp} (H)$ have symmetrical maxima of similar amplitudes, and they are characterized by almost the same slope $dT_c/dH$. In our opinion, this indicates that the nucleation of superconductivity, responsible for an initial deviation of the electrical resistance from its normal value, is almost isotropic (i.e. independent on the direction in which the bias current was injected and the voltage drop was recorded). Taking the position of the $T_c$ maxima and comparing the slope $dT_c/dH$ with that for the regime of surface superconductivity $dT_{c\perp}/dH \approx 0.6 T_{c0}/H_{c2}^{(0)}$, we estimate the amplitude of the nonuniform field $B_0 \approx 52$ mT, the upper critical field $\mu_0 H_{c2}^{(0)} \approx 20.4$ mT at $T = 0$, and the maximal critical temperature $T_{c0} \approx 1.35$ K. These values appear to be typical for pure Al films and bridges.

However deeper in the superconducting state in the $H$–$T$ plane the transport properties of the S/F hybrid system become essentially anisotropic. Figure 3 illustrates this, showing the dependencies of the resistance $R_{||} = V_{||}/I$ (top row) and $R_{\perp} = V_{\perp}/I$ (bottom row) as a function of $H$ and $I$, derived from the isothermal $I$–$V$ curves at constant $H$ value. As expected the total resistance of the sample goes to zero only for the parallel geometry when $I$ flows along domain walls [Fig. 3(a)–(c)]. Indeed, the stripe-type domain structure allows to form a continuous path for the supercurrents at $|H| \approx B_0$, connecting the electrodes I and II. It is easy to see that the maximal critical current corresponds to the most effective compensation, when one part of the bridge is subjected to zero local magnetic field, $B_z = \mu_0 H + b_z \approx 0$, while $B_z \approx 2B_0$ induces the normal state in the other part. Taking $I_{max} = 1035$ $\mu$A and the sample’s cross-section $S = 1.5 \cdot 10^{-8}$ cm$^2$, one can estimate the critical current density $j_c = 2I_{max}/S \approx 1.4 \cdot 10^7$ A/cm$^2$ at $T = 500$ mK, which can be interpreted as the depinning current density. Apparently, an increase of temperature reduces the size of the area of zero resistance in the $H$–$I$ plane. By contrast, alternating superconducting and normal (N) regions, induced by the magnetic template, act as a series of resistors if $I$ is injected perpendicular to the S-N interfaces. Consequently, the formation of superconductivity above the reverse domains roughly halves the total resistance of the sample at low temperatures: $\min V_{\perp}/I \approx R_n/2$ [Fig. 3(d)–(f)]. Therefore, in the vicinity of the compensation fields, we observe

![FIG. 2: (color online) The phase transition lines $T_c(H)$ estimated according to the criterium $V(H,T_c)/I_0 = 0.8 R_n$ for the measurements of magnetotransport using the contacts I and II (along the domain wall, red circles) and the contacts III and IV (across the domain walls, blue squares). $I_0 = 100$ $\mu$A is the dc bias current and $R_n$ is the normal state resistance.](image-url)
FIG. 3: (Color online) The dc resistance $R$ of the superconducting bridge as a function of the external magnetic field $H$ and the biased dc current $I$, measured along the domain wall [panels (a)–(c)] and across the domain wall [panels (d)–(f)]. (a) and (d) $T = 500$ mK, (b) and (e) $T = 900$ mK, (c) and (f) $T = 1300$ mK. Solid black lines are the curves of constant resistance: $R(H, I) = 0.5, 1.0, 1.5, 2.0, 2.5, 3.0, 3.5, 4.0$ and $4.5 \text{ Ohm}$. Note that the color scales of all plots are identical.

a giant anisotropy of resistivity: $\min R_{\perp}/\min R_{\parallel} > 10^3$ (Fig. 4), which is in agreement with that obtained for a network of parallel magnetic domains in permalloy films [17, 18, 19, 20].

In summary, we demonstrated that a 1D domain structure in a ferromagnetic substrate can induce a giant anisotropy of the electrical transport of S films that are placed on top of the substrate. This effect is caused by the appearance of superconducting channels that run along the underlying magnetic domains. We also studied the $H$- and $T$ dependence of the critical current through such an individual channel.
FIG. 4: (Color online) Giant resistance anisotropy illustrated by cross-sections of $R_{\parallel}(H, I)$ [panel (a) in Fig. 3] and $R_{\perp}(H, I)$ [panel (d) in Fig. 3] taken at $T = 500$ mK and $I = 600 \mu$A (a) and $I = 100 \mu$A (b). Red circles (blue squares) correspond to the resistivity along (across) the domain wall.

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[1] A.I. Buzdin, Rev. Mod. Phys. 77, 935 (2005).
[2] I.F. Lyuksyutov and V.L. Pokrovsky, Adv. Phys. 57, 67 (2005).
[3] M. Vélez, J.I. Martín, J.E. Villegas, A. Hoffmann, E.M. González, J.L. Vicent, and I.K. Schuller, Journ. Magn. Magn. Mater. 320, 2547–2562 (2008).
[4] A.Yu. Aladyshkin, A.V. Silhanek, W. Gillijns, and V.V. Moshchalkov, Supercond. Sci. Tech. 22, 053001 (2009).
[5] M. Lange, M.J. Van Bael, Y. Bruynseraede, and V.V. Moshchalkov, Phys. Rev. Lett. 90, 197006 (2003).
[6] A.Yu. Aladyshkin, A.I. Buzdin, A.A. Fraerman, A.S. Melnikov, D.A. Ryzhov, A.V. Sokolov, Phys. Rev. B 68, 184508 (2003).
[7] Z. Yang, M. Lange, A. Volodin, R. Szmyczak, and V.V. Moshchalkov, Nature Mater. 3, 793 (2004).
[8] W. Gillijns, A.Yu. Aladyshkin, A.V. Silhanek, and V.V. Moshchalkov, Phys. Rev. B 76, 060503(R) (2007).
[9] D.J. Morgan and J.B. Ketterson, Phys. Rev. Lett. 80, 3614(1998).
[10] M.J. Van Bael, K. Temst, V.V. Moshchalkov, and Y. Bruynseraede, Phys. Rev. B 59, 14674 (1999).
[11] M. Lange, M.J. van Bael, V.V. Moshchalkov, and Y. Bruynseraede, Appl. Phys. Lett. 81, 322 (2002).
[12] J.I. Martín, M. Vélez, J. Nogués, and I.K. Schuller, Phys. Rev. Lett. 79, 1929 (1997).
[13] Y. Jaccard, J.I. Martín, M.-C. Cyrille, M. Vélez, J.L. Vicent, and I.K. Schuller, Phys. Rev. B 58, 8232 (1998).
[14] A. Hoffmann, P. Prieto, and I.K. Schuller, Phys. Rev. B 61, 6958 (2000).
[15] D.Y. Vodolazov, B.A. Gribkov, S.A. Gusev, A.Yu. Klimov, Yu.N. Nozdrin, V.V. Rogov, and S.N. Vdovichev, Phys. Rev. B 72, 064509 (2005).
[16] M. Morelle and V.V. Moshchalkov, Appl. Phys. Lett. 88, 172507 (2006).
[17] V. Vlasko-Vlasov, U. Welp, G. Karapetrov, V. Novosad, D. Rosenmann, M. Iavarone, A. Belkin, and W.-K. Kwok, Phys. Rev. B 77, 134518 (2008).
[18] A. Belkin, V. Novosad, M. Iavarone, J. Fedor, J.E. Pearson, A. Petrean-Troncalli, and G. Karapetrov, Appl. Phys. Lett. 93, 072510 (2008).
[19] A. Belkin, V. Novosad, M. Iavarone, J. Pearson, and G. Karapetrov, Phys. Rev. B 77, 180506 (2008).
[20] V.K. Vlasko-Vlasov, U. Welp, A. Imre, D. Rosenmann, J. Pearson, and W.K. Kwok, Phys. Rev. B 78, 214511 (2008).
[21] M.V. Milošević, G. Berdiyorov, and F.M. Peeters, Phys. Rev. Lett. 95, 147004 (2005).