Electronic structures of WS$_2$ armchair nanoribbons doped with transition metals

Yan-Hong Chen$^1$, Chi-Hsuan Lee$^2$, Shun-Jen Cheng$^1$ & Chih-Kai Yang$^{1*}$

Armchair WS$_2$ nanoribbons are semiconductors with band gaps close to 0.5 eV. If some of the W atoms in the ribbon are replaced by transition metals, the impurity states can tremendously affect the overall electronic structure of the doped ribbon. By using first-principles calculations based on density functional theory, we investigated substitutional doping of Ti, V, Cr, Mn, Fe, and Co at various positions on WS$_2$ ribbons of different widths. We found that Fe-doped ribbons can have two-channel conduction in the middle segment of the ribbon and at the edges, carrying opposite spins separately. Many Co-doped ribbons are transformed into spin filters that exhibit 100% spin-polarized conduction. These results will be useful for spintronics and nanoelectronic circuit design.

Results and discussion

Intrinsic armchair WS$_2$ nanoribbons. We first present energy bands of intrinsic Na-WS$_2$ nanoribbons, where Na is 15, 20, or 25, representing ribbons of three different widths. Each WS$_2$ ribbon [Na = 15, 20, and 25 in Fig. 1a–c, respectively] manifests its finite width by valence and conduction bands mainly associated with W atoms at its armchair edges, including a flat band (Fig. 1a–c) at approximately 0.6 eV. The contribution from the S atoms at the edges also forms a nearly flat band that is much lower in energy and mainly within the energy

---

$^1$Department of Electrophysics, National Chiao Tung University, Hsinchu, Taiwan, ROC. $^2$Graduate Institute of Applied Physics, National Chengchi University, Taipei, Taiwan, ROC. $^*$email: ckyang@nccu.edu.tw
bands of the bulk. The band gap is 0.468 eV for 15-WS₂, increases to 0.498 eV for 20-WS₂, and then stays almost unchanged for the even wider 25-WS₂.

**Doping at the center of 15-WS₂.** We next discuss the configuration in which the substitutional doping occurs in the central position of the 15-WS₂ ribbon and a W atom is replaced by a Ti atom at a concentration of 1/15 (6.67%). Figure 2 plots the exact position of the substitution and calculated electronic structure. Compared with the pristine 15-WS₂, the most striking features are three (with two almost degenerate) energy bands crossing the Fermi level $E_F$, rendering the centrally doped Ti-15-WS₂ structure a conductor. The band with the highest energy of the three is almost exclusively composed of the $d$ orbitals from the Ti impurity and its nearest neighbors of W atoms, which are marked collectively by green spheres in Fig. 2b with sizes proportional to their contributions, and also curves of local density of states (LDOS) in partial waves in Fig. 2c,d. The other two bands are likewise derived heavily from the $d$ orbitals of the Ti, but a conspicuous presence of edge W atoms (as indicated by brown spheres in Fig. 2b and the corresponding LDOS in Fig. 2e) also emerges, largely below $E_F$.

We therefore established that conduction of the centrally doped Ti-15-WS₂ ribbon at a concentration of 6.67% is overwhelmingly carried out by the Ti and its neighboring W atoms, whereas the basic architecture of the pure WS₂ ribbon energy bands consisting of edge W and S atoms remains largely intact. The wave function of Ti does not have an appreciable overlapping with those of W atoms at both edges. Analysis of the $d$ orbitals by partial waves further identifies the three major constituents in $d_{xy}, d_{x^2−y^2}$, and $d_{z^2}$.

Many of the descriptions of the centrally doped Ti-15-WS₂ can also be applied to V-15-WS₂. The dopant V, however, has one more electron in its $d$ shells, or one hole less as compared with Ti. Implied in the rigid band estimate is that fewer bands would cross $E_F$ as a result. This is exactly what the density functional calculation shows in Fig. 3a, in which only two bands cross $E_F$ and most of their electronic states are below $E_F$. The reduction in the available states also proceeds through the impurity and its surrounding W atoms, where the wave functions have the same $d$ components just as those in the Ti case.

Having exactly the same electrons in the $d$ shell as W, Cr as a dopant is not expected to alter the electrical property of the pure WS₂ ribbon. The bands contributed by Cr either stay well below $E_F$ or are pushed high above $E_F$, with little mixing between the orbitals of Cr and W atoms at the edges (Fig. 3b). Centrally doped Cr-15-WS₂ remains a semiconductor.

Ti, V, and Cr-doped WS₂ ribbons are essentially non-magnetic, but doping with Mn, Fe, and Co, classified as N-type, renders the ribbon magnetic. Using spin-polarized calculation, we are able to separate the energy bands corresponding to the majority spin (Fig. 4a) from those of the minority spin (Fig. 4b), for the 15-WS₂ doped with a Mn in the center. Figure 4a clearly identifies two bands associated with the majority spin approaching $E_F$ from above and below separately. The former is contributed by edge W atoms and the latter mainly by the Mn and its neighboring W atoms. The two curves are flat when they are closest to the Fermi level and it is likely that these states will not contribute to conduction in an electronic transport setup. No bands of the minority spin come so close to $E_F$ in Fig. 4b. The imbalance of the distribution of spins produces a magnetic moment of 0.938 μB per unit cell for the doped ribbon. To further clarify the band structure near $E_F$, we also undertook calculations that incorporated the spin–orbit interaction. The result supports the idea of a semiconductor with an indirect band gap of approximately 60 meV (Fig. 4c).

Proceeding next to Fe doping, which has one more electron available per unit cell, the electronic structure is very different from the previous examples. The 15-WS₂ with a centrally doped Fe is now a conductor with two...
Figure 2. (a) Unit cell and (b) energy bands of Ti-15-WS$_2$. LDOS in partial waves of (c) Ti, (d) W atoms closest to Ti, and (e) W at the edges.

Figure 3. Energy bands of centrally doped (a) V-15-WS$_2$ and (b) Cr-15-WS$_2$. 
bands crossing $E_F$. Each band is associated with the conductance of one quantum unit $e^2/h$ in a ballistic regime for a perfect periodic system, where $e$ is the electron charge and $h$ is Planck constant. The band with the majority spin (Fig. 5a) is comprised of the orbitals mainly from Fe and its surrounding W, indicating available states in the central region of the ribbon. The band of the minority spin (Fig. 5b) is strongly associated with W atoms at both edges. Mixing of wave functions between Fe and the edges generally occurs at energies greater than $E_F$. Thus, even though charge transport can occur both in the interior and edges of the ribbon, the associated spins are separately carried by electrons in the two channels, with the former bearing the majority spin and the latter the minority spin. The entire unit cell of the ribbon exhibits a magnetic moment of 1.128 $\mu_B$. A further calculation taking into account the spin–orbit interaction also confirms this two-channel transport scenario (Fig. 5c). Figure 5d shows the distribution of spin density, where the majority spin occupying the central region of the ribbon is clearly separated from the minority spin confined at the edges.
Continuing the study of TM-doped 15-WS$_2$ nanoribbons, we now substitute a Co for the W atom at the central position of the unit cell. The calculated band structure is surprising in terms of its physical significance. There are now two bands crossing $E_F$, both exclusively associated with the majority spin (Fig. 6a), but there are none doing so for the minority spin (Fig. 6b). The ribbon is therefore transformed into a spin filter, allowing only passage of electrons that have the majority spin.

From partial wave analysis of the LDOS (Fig. 7), it is easy to establish that the two bands of the majority spin crossing $E_F$ are formed by $d$ orbitals of Co and the W atoms nearby, and negligibly hybridize with edge atoms in the context of conduction.

Dopants closer to one side of 15-WS$_2$. If the doped impurity atom is placed closer to one side of the nanoribbon, its wave function is expected to overlap more with those of the edge atoms on that side, and the resulting hybridization can significantly affect the electronic structure. In Fig. 8a, six sites away from the center of the 15-WS$_2$ nanoribbon and numbered from 1–6 are marks for substitutional doping of the TM atoms. A larger number corresponds to a dopant that is closer to the left edge. Figure 8b plots the binding energies $E_B$ for each dopant at the six sites and the central position (marked by 0). $E_B$ is defined as the difference between the total energy of the doped structure $E_{\text{total}}$(ribbon − W + dopant) and the sum of the total energies of the structure with a W vacancy $E_{\text{total}}$(ribbon − W) and the free dopant $E_{\text{total}}$(dopant), or $E_B = E_{\text{total}}$(ribbon − W + dopant) − [$E_{\text{total}}$(ribbon − W) + $E_{\text{total}}$(dopant)]. It is clear that Ti is most tightly bound to the 15-WS$_2$ nanoribbon, and each of the TM atoms is more likely to stay close to the center of the ribbon than the edges.

We also calculated average formation energy $E_{\text{form}}$ for each of the configurations in Fig. 8b. $E_{\text{form}}$ was derived by calculating the difference between $E_{\text{total}}$(ribbon − W + dopant) and the sum of total energies of its constituents in isolation and then dividing the difference by the total number of atoms in the system. The results are plotted in Fig. 8c.

Continuing with Co-15-WS$_2$, the structure remains a 100% spin polarizer with Co occupying position 1, 2, or 3. Figure 9a demonstrates that only energy bands associated with the majority spin cross $E_F$ for the first three doping sites. If Co is placed in positions even closer to the left edge, the larger overlapping of wave functions between Co and W at the left edge significantly changes the energy bands. At positions 4 and 6, band gaps are opened and the ribbon becomes a semiconductor. Although a single band does cross $E_F$ for the dopant at position 5, the band carries both spins and is not a spin filter. As expected, the bands responsible for conduction of the doped structure with the dopant Co at position 1, 2, 3, or 5 consist of $d$ orbitals of Co (Fig. 9b), LDOS], and its W neighbors. Part I of Supplementary Information provides a detailed illustration of the evolution of the LDOS for Co at various positions and W atoms at both edges. In Fe-15-WS$_2$, the Fe atom at position 1 switches the structure from a conductor to a semiconductor (Fig. 10a,b). The shortest displacement from the central position is enough to break the symmetry and induces splitting of the energy bands. Combined with the mixing of wave functions between the off-centered dopant and the W atoms at the nearer edge, a zero-gap semiconductor is created. Two-channel conduction, however, is maintained for Fe at positions 2 or 3, where transport of electrons with the majority spin is still well inside the ribbon, whereas movement of electrons with the minority spin is now limited to the left edge only, which is now

![Figure 6. Energy bands of Co-15-WS$_2$ for the (a) majority spin and (b) minority spin.](image-url)
Figure 7. (a) Unit cell and (b) energy bands with the majority spin of Co-15-WS₂. LDOS of the majority spin in partial waves of (c) Co, (d) W atoms closest to Co, and (e) W at the edges.

Figure 8. (a) Doping sites marked by numbers. (b) Binding energy for each of the six TM dopants at marked positions. (c) Formation energy for each of the possible configurations.
closer to the dopant. Figure 10c shows the LDOS of the minority spin for W at the left edge to display the evolving distribution of $d$ orbitals. The wave function of Fe at position 4, 5, or 6 is so heavily hybridized with that of W at the left edge, the LDOS curves bear almost no resemblance to the undisturbed edge states. For Fe at position 4 or 6, the structure is a semiconductor, whereas at position 5 a conductor emerges that accommodates both spins. Similar transformations between metals and semiconductors also occur for other TM dopants placed closer to one side of the nanoribbon with the exception of Cr. Cr-15-WS$_2$ is always a semiconductor regardless of the position of the Cr impurity.

Lower concentration of dopants. It is possible to reduce the interaction between the dopant and edge atoms by widening the width of the WS$_2$ nanoribbon. This is a method of reducing the dopant concentration. However, it should be cautioned that the calculation is still within the periodic framework and the system is highly ordered no matter how low the concentration is. We chose two widths designated as Na 20 and 25 for further study of the TM-doped system, with the dopant inevitably closer to one side of the ribbon in the former system and still occupying the central position in the latter. Ti and V-doped wider nanoribbons remain conductors through charge carriers of the dopants and their neighboring W atoms. Cr-doped ribbons are still semiconductors. Mn-doped ribbons are also semiconductors, all with small band gaps.

The most striking change occurs on Fe. As the width of the ribbon is widened, the energy bands contributed by the W atoms at the edges are raised and eventually above $E_F$ and out of conduction. This is mainly because of less hybridization between Fe and W atoms at the edges. Fe-25-WS$_2$, for example, no longer provides two-channel conduction and effectively becomes a perfect spin polarizer by only passing electrons with the majority spin. Part II of Supplementary Information shows the evolution of spin density from Fe-15-WS$_2$ to Fe-25-WS$_2$. 

![Figure 9](https://www.nature.com/scientificreports/)
Figure 10. Energy bands of the (a) majority spin and (b) minority spin for Fe doping at positions from 1–6. (c) LDOS of the minority spin for W at the left edge with Fe occupying positions from 1–6 (from left to right).
and clearly indicates that the spin density at the edges of Fe-25-WS₂ vanishes. Partial wave analysis of its energy bands (Fig. 11) also assists in establishing this picture.

Both Co-20-WS₂ and Co-25-WS₂ are also spin filters with larger ribbon widths. Two bands that are associated with the majority spin crossing $E_F$ are a feature common to both Co-doped ribbons (Fig. 12a,b), which suggests a reliable method of constructing a spin-polarized circuit through Co doping. One question arises whether a continuing Co chain (Fig. 13a, 15-WS₂) is also a spin filter. With two bands separately belonging to different spins
crossing $E_F$ (Fig. 13b), the structure—with 2× the Co concentration and thus strengthened interaction between the dopants—becomes an ordinary conductor.

Another approach for evaluating the effect of doping at a lower concentration is to increase the distance between the dopants along the ribbon length. If the distance between the dopants is doubled in the 15-WS$_2$ ribbon, the concentration decreases to 3.33% and interaction between the dopants is expected to be considerably reduced. As it turns out, the lower concentration of dopants flattens their associated energy bands. Ti and V-doped structures are still conductors and the Cr-doped ribbon continues to have a band gap. Dopants of Mn, Fe, and Co behave quite differently, however, Mn-15-WS$_2$ at this low concentration becomes a conductor, whereas Fe and Co each opens a small band gap (Fig. 14).

In summary, we showed that the electronic structure of an armchair WS$_2$ nanoribbon can be tuned by substitutional doping of TM impurities. The dopant concentration, ribbon width, and dopant position all contribute to the doped nanoribbon's transformation between a semiconductor and metal. Among the dopants, Fe-15-WS$_2$ is capable of supporting two-channel conduction, and wider nanoribbons such as Fe-25-WS$_2$ are perfect spin filters. Co-15-WS$_2$ with the dopant closer to the center and corresponding wider versions also consistently support conduction of 100% spin polarization. These results will be useful for spintronics and designing nanoelectronic circuits in a broad range of applications.

Figure 13. (a) Unit cell and (b) energy bands for a continuous Co chain on 15-WS$_2$.

Figure 14. Energy bands of 15-WS$_2$ for (a) Mn, (b) Fe, and (c) Co doping at a lower concentration of 3.33%.
Methods
In deriving the optimal configurations of nanoribbons and energy bands, we performed spin-polarized density functional calculations using the VASP code\(^{23,34}\). Projector augmented-wave pseudopotentials and the exchange–correlation functionals of Perdew, Burke and Ernzerhof\(^{34}\) were chosen to execute the calculation. A 1 × 81 × 1 Monkhorst–Pack k-point mesh was used for sampling k points in the first Brillouin zone. We allocated a length of 20 Å in two perpendicular directions for vacuum space to eliminate artificial interaction between the supercells containing the ribbon. The cut-off energy for the expansion of the wave functions and potentials in the plane-wave basis was 400 eV.

Received: 9 December 2019; Accepted: 26 August 2020
Published online: 05 October 2020

References
1. Novoselov, K. S. et al. Two-dimensional atomic crystals. Proc. Natl. Acad. Sci. USA 102, 10451–10453 (2005).
2. Eda, G., Yamaguchi, H., Voiry, D., Chen, M. & Chhowalla, M. Photoluminescence from chemically exfoliated MoS\(_2\). Nano Lett. 11, 5111–5116 (2011).
3. Li, H., Wu, J., Yin, Z. & Zhang, H. Preparation and applications of mechanically exfoliated single-layer and multilayer MoS\(_2\) and WSe\(_2\) nanosheets. Acc. Chem. Res. 47, 1067–1075 (2014).
4. Samad, L. et al. Layer-controlled chemical vapor deposition growth of MoS\(_2\) vertical heterostructures via van der Waals epitaxy. ACS Nano 10, 7039–7046 (2016).
5. Yu, Y. et al. Controlled scalable synthesis of uniform, high-quality monolayer and few-layer MoS\(_2\) films. Sci. Rep. 3, 1866 (2013).
6. Jurca, T. et al. Low-temperature atomic layer deposition of MoS\(_2\) films. Angew. Chem. Int. Ed. 56, 4991–4995 (2017).
7. Tang, H. et al. Highly scalable synthesis of MoS\(_2\) thin films with precise thickness control via polymer-assisted deposition. Chem. Mater. 29, 5772–5776 (2017).
8. Li, S. et al. Vapor-liquid-solid growth of monolayer MoS\(_2\) nanoribbons. Nat. Mater. 17, 535–542 (2018).
9. Wang, Z. et al. Mixed low-dimensional nanomaterial: 2D ultranarrow MoS\(_2\) inorganic nanoribbons encapsulated in quasi-1D carbon nanotubes. J. Am. Chem. Soc. 132, 13840–13847 (2010).
10. Lin, J. et al. Enhanced electrocatalysis for hydrogen evolution reactions from WS\(_2\) nanoribbons. Adv. Energy Mater. 4, 1301875 (2014).
11. Mak, K. F., Lee, C., Hone, J., Shan, J. & Heinz, T. F. Atomically thin MoS\(_2\): a new direct-gap semiconductor. Phys. Rev. Lett. 105, 136805 (2010).
12. Lee, C. et al. Anomalous lattice vibrations of single- and few-layer MoS\(_2\). ACS Nano 4, 2695–2700 (2010).
13. Splendiani, A. et al. Emerging photoluminescence in monolayer MoS\(_2\). Nano Lett. 10, 1271–1275 (2010).
14. Radisavljevic, B., Radenovic, A., Brivio, J., Giacometti, V. & Kis, A. Single-layer MoS\(_2\) transistors. Nat. Nanotechnol. 6, 147–150 (2011).
15. Xu, X., Yao, W., Xiao, D. & Heinz, T. F. Spin and pseudospins in layered transition metal dichalcogenides. Nat. Phys. 10, 343–350 (2014).
16. Gibertini, M. & Marzari, N. Emergence of one-dimensional wires of free carriers in transition-metal-dichalcogenide nanostructures. Nano Lett. 15, 6229–6238 (2015).
17. Heyne, M. H. et al. Two-dimensional WS\(_2\) nanoribbon deposition by conversion of pre-patterned amorphous silicon. Nanotechnology 28, 04LT01 (2017).
18. Wang, Z. et al. Ultra-narrow WS\(_2\) nanoribbons encapsulated in carbon nanotubes. J. Mater. Chem. 21, 171–180 (2011).
19. Khalil, A. et al. Metallic 1T-WS\(_2\), nanoribbons as highly conductive electrodes for supercapacitors. RSC Adv. 6, 48788–48791 (2016).
20. Zhang, H., Li, X.-B. & Liu, L.-M. Tunable electronic and magnetic properties of WS\(_2\) nanoribbons. J. Appl. Phys. 114, 093910 (2013).
21. López-Urías, F. et al. Electronic, magnetic, optical, and edge-reactivity properties of semiconducting and metallic WS\(_2\) nanoribbons. 2D Mater. 2, 015002 (2015).
22. Xiao, S.-L., Yu, W.-Z. & Gao, S.-P. Edge preference and band gap characters of Mo\(_2\)S\(_2\) and WS\(_2\) nanoribbons. Surf. Sci. 653, 107–112 (2016).
23. Lin, J. et al. Enhanced electrocatalysis for hydrogen evolution reactions from WS\(_2\) nanoribbons. Adv. Energy Mater. 4, 1301875 (2014).
24. Andriotis, A. N. & Menon, M. Tunable magnetic properties of transition metal doped MoS\(_2\). Phys. Rev. B 90, 125304 (2014).
25. Wang, Y., Li, S. & Yi, J. Electronic and magnetic properties of Co doped Mo\(_2\)S\(_2\) monolayer. Sci. Rep. 6, 24153 (2016).
26. Lin, X. & Ji, J. Charge and magnetic states of Mn-, Fe-, and Co-doped monolayer MoS\(_2\). J. Appl. Phys. 116, 044311 (2014).
27. Lin, Y. et al. Revealing the atomic defects of WS\(_2\) governing its distinct optical emissions. Nano Lett. 14, 39248–39253 (2014).
28. Pedley, N., Kumar, A. & Chakrabarti, S. First principle study of temperature-dependent magnetoresistance and spin filtration effect in WS\(_2\) nanoribbon. ACS Appl. Mater. Interfaces 11, 39248–39253 (2019).
29. Pedstone, A. A., Lewis, D. J. & O’Brien, P. Synthesis, properties, and applications of transition metal-doped layered transition metal dichalcogenides. Chem. Mater. 28, 1965–1974 (2016).
30. Ramasubramaniam, A. & Naveh, D. Mn doped monolayer MoS\(_2\): An atomically thin dilute magnetic semiconductor. Phys. Rev. B 87, 195201 (2013).
31. Andriotis, A. N. & Menon, M. Tunable magnetic properties of transition metal doped MoS\(_2\). Phys. Rev. B 90, 125304 (2014).
32. Wang, Y., Li, S. & Yi, J. Electronic and magnetic properties of Co doped Mo\(_2\)S\(_2\) monolayer. Sci. Rep. 6, 24153 (2016).
33. Lin, X. & Ji, J. Charge and magnetic states of Mn-, Fe-, and Co-doped monolayer MoS\(_2\). J. Appl. Phys. 116, 044311 (2014).
34. Kresse, G. & Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. Phys. Rev. B 54, 11169–11186 (1996).
35. Kresse, G. & Furthmüller, J. Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. Comput. Mater. Sci. 6, 15–50 (1996).
36. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. Phys. Rev. Lett. 77, 3865–3868 (1996).

Acknowledgements
This work was supported by the Ministry of Science and Technology of the Republic of China under Grant Numbers MOST 109–2639–E–009–001, MOST 109–2112–M–009–018–MY3, MOST 108–2112–M–004–001, and MOST 109–2112–M–004–002. Supports from the National Centers for Theoretical Sciences and High-Performance Computing of the ROC are also gratefully acknowledged.
Author contributions
C.K.Y. and S.J.C. initiated and supervised the research. Y.H.C. and C.H.L. performed the calculations. C.K.Y. wrote the manuscript. All authors reviewed the manuscript.

Competing interests
The authors declare no competing interests.

Additional information
Supplementary information is available for this paper at https://doi.org/10.1038/s41598-020-73602-2.

Correspondence and requests for materials should be addressed to C.-K.Y.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher’s note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons licence, and indicate if changes were made. The images or other third party material in this article are included in the article’s Creative Commons licence, unless indicated otherwise in a credit line to the material. If material is not included in the article’s Creative Commons licence and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this licence, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2020