Enhancing the mass sensitivity of graphene nanoresonators via nonlinear oscillations: the effective strain mechanism

Jin-Wu Jiang¹, Harold S Park² and Timon Rabczuk¹

¹ Institute of Structural Mechanics, Bauhaus-University Weimar, Marienstraße 15, D-99423 Weimar, Germany
² Department of Mechanical Engineering, Boston University, Boston, MA 02215, USA

E-mail: parkhs@bu.edu and timon.rabczuk@uni-weimar.de

Received 7 July 2012, in final form 22 September 2012
Published 1 November 2012
Online at stacks.iop.org/Nano/23/475501

Abstract
We perform classical molecular dynamics simulations to investigate the enhancement of the mass sensitivity and resonant frequency of graphene nanomechanical resonators that is achieved by driving them into the nonlinear oscillation regime. The mass sensitivity as measured by the resonant frequency shift is found to triple if the actuation energy is about 2.5 times the initial kinetic energy of the nanoresonator. The mechanism underlying the enhanced mass sensitivity is found to be the effective strain that is induced in the nanoresonator due to the nonlinear oscillations, where we obtain an analytic relationship between the induced effective strain and the actuation energy that is applied to the graphene nanoresonator. An important implication of this work is that there is no need for experimentalists to apply tensile strain to the resonators before actuation in order to enhance the mass sensitivity. Instead, enhanced mass sensitivity can be obtained by the far simpler technique of actuating nonlinear oscillations of an existing graphene nanoresonator.

(Some figures may appear in colour only in the online journal)

1. Introduction

Carbon nanotube and graphene-based nanomechanical resonators (GNMRs) have attracted significant attention from the scientific community; see the recent reviews [1–4]. Experiments have demonstrated that single-walled carbon nanotube nanomechanical resonators can serve as mass sensors that are capable of detecting individual atoms or molecules [5–7], which is made possible by the high stiffness and low mass of carbon nanotubes. Graphene also possesses high stiffness [8–10], but may prove to be a superior mass sensor to nanotubes due to its significantly larger surface area to which molecules can attach.

The performance of a GNMR mass sensor is closely related to its quality (Q)-factor, which reflects the energy that is lost during the mechanical oscillation of the resonator. In the first reported study of GNMRs, Bunch et al observed very low (<100) Q-factors for GNMRs working in the megahertz range [11]. In a later experiment, the same group reported a dramatic increase of the Q-factor at lower temperatures, with the Q-factors reaching up to 9000 at 10 K for GNMRs produced by the chemical vapor deposition growth method [12]. Chen et al also found that the Q-factor of a GNMR increases with decreasing temperature, and reaches 10⁴ at 5 K [13]. More recently, Eichler et al [14] found that the Q-factor of a GNMR can reach a value of 10⁵, which is very close to the theoretical Q-factor upper bound of about 10⁶, which has been predicted for GNMRs with all edges fixed coupled with idealized mechanical actuation [15, 16].

Because a key sensing objective for GNMRs is to enable detection of individual molecules or atoms, it is critical to determine methods of enhancing their mass sensitivity. For
example, Eichler et al have shown that it is possible to increase the resonant frequency (and thus the mass sensitivity) of a GNMR by driving its mechanical oscillation into the nonlinear regime; this was explained using a continuum mechanics model [14]. Using similar continuum models, several groups have theoretically studied the nonlinear effect on the mass sensitivity of nanoresonators [17–19], with only a single, recent study related to graphene [20]. Besides the nonlinear resonance, mechanical strain is another possible method to improve the mass sensitivity of a GNMR, as the resonant frequency can be enhanced by mechanical tension [21, 23, 18, 22, 24].

In this paper, we investigate the utility of inducing nonlinear oscillations to enhance the mass sensitivity of GBMs using a classical molecular dynamics (MD) simulation. Our simulations show that the adsorption-induced frequency shift resulting from a single Au atom can be enhanced by increasing the actuation energy when the actuation energy parameter \( \alpha \) is below a critical value, \( \alpha_c = 2.0875 \pm 0.0125 \). For actuation energies above the critical value, the adsorbed Au atom exhibits significant diffusion, which reduces the \( Q \)-factor of the GNMR by more than three orders of magnitude and significantly degrades the mass sensitivity. Quite differently from the widely used continuum mechanics explanation, we show that the nonlinear-induced frequency enhancement results from the ‘effective strain’ that is induced in the oscillating GNMR. We derive an analytic expression for the effective strain, \( \epsilon_\alpha = \frac{3}{2} \pi^2 \alpha \frac{E_0^0}{m \omega \sqrt{L_x L_y}} \) (here \( E_0^0 \) is the kinetic energy, \( m \) the mass, \( \omega \) the angular frequency and \( L \) the length of the resonator), that enables us to directly link the equivalence of the applied mechanical tensile strain and the strain induced by nonlinear oscillations to the resonant frequencies of GNMRs.

2. Simulation details

The graphene sample in our simulations has dimensions \((L_x, L_y) = (34, 38) \text{ Å}\) and is composed of 504 carbon atoms. The atoms at the +\( x \) and −\( x \) ends of the GNMR are fixed, while periodic boundary conditions are applied in the \( y \) direction. The interactions of the carbon atoms are described by the Brenner (REBO-II) potential [25]. For the cases where a single Au atom is adsorbed on the GNMR, the interaction between the Au atom and the GNMR is modeled by a Lennard-Jones potential with length parameter \( \sigma = 2.9943 \text{ Å} \) and energy parameter \( \epsilon = 0.02936 \text{ eV} \) [22]. The standard Newton equations of motion are integrated in time using the velocity Verlet algorithm with a time step of 1 fs.

Our simulations are performed as follows. First, the Nosé–Hoover [26, 27] thermostat is applied to thermalize the system to a constant temperature of 4.2 K within the NVT (i.e. the particles number \( N \), the volume \( V \) and the temperature \( T \) of the system are constant) ensemble, which is run for \( 10^5 \) MD steps. The mechanical oscillation of the resonator is then actuated by adding a velocity distribution to the system, which follows the morphology of the first flexural vibrational mode of graphene [16]. The imposed velocity distribution, or actuation energy, is \( \Delta E = \alpha E_0^0 \), where \( E_0^0 \) is the total kinetic energy in the GNMR after thermalization but just before its actuation and \( \alpha \) is the actuation energy parameter. After the actuation energy is applied, the system is allowed to oscillate freely within the NVE (i.e. the particles number \( N \), the volume \( V \) and the energy \( E \) of the system are constant) ensemble for \( 21^9 \) MD steps. The data from the NVE ensemble are used to analyze the mechanical oscillation of the GNMR.

3. Results

During the free vibration period, the energy in the GNMR switches between kinetic and potential energy. The frequency of the switching is \( 2f \), with \( f \) being the frequency of mechanical oscillation. Figure 1 shows the resonant peaks that are obtained by taking a Fourier transformation of the time history of the kinetic energy; the resonant peaks are used to extract the resonant frequency \( f \) at liquid helium temperature (4.2 K) for GNMRs with different actuation energies \( \alpha = 0.1, 0.5, 1.0, 1.5 \) and 2.0, where, for reference, \( \alpha = 1 \) means that the actuation energy equals the total kinetic energy in the GNMR after thermalization. It should be noted that the amplitude in figure 1 is large, because of the long simulation time we have used. We focus on the liquid helium temperature because this temperature is commonly utilized in experiments involving GNMRs (e.g. 90 mK or 4.0 K in [14]).

Figure 2 shows the resonant frequency as a function of actuation energy \( \alpha \). Panel (a) shows that the resonant frequency is enhanced by increasing \( \alpha \). This enhancement is due to the increasingly nonlinear behavior of the GNMR induced by the increasing oscillation amplitude. Adsorption of a single Au atom on top of the GNMR causes a considerable reduction of the resonant frequency, due to the resulting increase in the effective mass of the GNMR. This frequency shift is what is measured experimentally to detect the adsorbed mass, and it is seen in panel (b) that the magnitude of the frequency shift increases by a factor of three for large actuation energies (\( \alpha \approx 2.5 \)) as compared to that if the GNMR was actuated with a small \( \alpha \). Panel (b) shows that the frequency shift can be increased by applying a larger actuation energy, which also results in increased mass sensitivity.
of the GNMR is greatly reduced, as shown in figure 5. There

\[ \alpha \]

energy, the mean free path is observed at a critical value of actuation

mechanical oscillation of the GNMR. As a result, the

noise provides important energy damping channels for the

for actuation energies above the critical value

\[ \alpha > \alpha_c \]

path for the Au atom as a function of \( \alpha \), which results from the diffusion of the adsorbate. In contrast, diffusion

occurs at \( \alpha = 2.0875 \pm 0.0125 \), implying diffusion of the Au atom for \( \alpha > \alpha_c \). The left top inset shows smooth resonant curves for

smaller actuation energies, \( \alpha \). These results also follow physical intuition. For example, diffusion happens at smaller \( \alpha \) for adsorbates that have a weaker bonding strength with graphene. In contrast, diffusion happens at larger \( \alpha \) for adsorbates with less mass, since these atoms have a smaller kinetic energy at the same temperature as larger mass adsorbates, and therefore more energy via the nonlinear actuation parameter \( \alpha \) is needed to induce diffusion for the smaller mass atoms. These results may serve as useful guidelines for experimentalists to verify our theoretical predictions, as different types of adsorbate (of different mass and/or different interaction strength with graphene) are usually observed experimentally.

However, as shown in the same figure, the frequency shift does not increase monotonically, and a decrease with increasing actuation energy is observed when \( \alpha \) is large.

The reduction in the frequency shift with increasing \( \alpha \) is determined to be a result of diffusion of the adsorbed Au atom, which was previously observed to occur at GNMR temperatures exceeding about 30 K [22]. To confirm the diffusion mechanism, we calculate in figure 3 the mean free path for the Au atom as a function of \( \alpha \), where a sharp jump in the mean free path is observed at a critical value of actuation energy, \( \alpha_c = 2.0875 \pm 0.0125 \). For \( \alpha < \alpha_c \), the mean free path of the Au atom is around 0.5 Å ps\(^{-1}\) with small fluctuations. In contrast, for \( \alpha > \alpha_c \), the mean free path increases to a value of around 1.0 Å ps\(^{-1}\), which implies diffusion of the Au atom.

The left top inset of figure 3 shows resonant curves for smaller actuation energies, \( \alpha = 0.1 \) and 1.5. These smooth curves provide evidence that the mechanical oscillation of the GNMR is the only vibrational mode in the system. The right bottom inset of figure 3 shows a significant amount of diffusion-induced noise in the resonant curves for large actuation energy parameters, \( \alpha = 4.0 \) and 10.0.

is almost no energy dissipation in the GNMR for actuation energies below \( \alpha_c \), leading to extremely high \( Q \)-factors. For actuation energies above the critical value, the \( Q \)-factor is reduced by more than three orders of magnitude. The time history of the kinetic energy is shown in figure 5(b) for \( \alpha = 2.075 \) and in figure 5(c) for \( \alpha = 2.1 \), from which the \( Q \)-factor has been extracted [16].

Figure 6 shows the \( Q \)-factor versus the actuation energy parameter \( \alpha \) for graphene nanomechanical resonators with three different adsorbates. In these three systems, the adsorbate mass \( m \) and the Lennard-Jones interaction parameter \( \epsilon \) are \((m, \epsilon)\), \((m/2, \epsilon)\) and \((m, \epsilon/2)\). For the first system \((m, \epsilon)\), the mass of the adsorbate is \( m = 197 \) and the Lennard-Jones potential parameter is \( \epsilon = 0.02936 \) eV, which are the actual values for the Au adsorbate. For the second system \((m/2, \epsilon)\), the mass of the adsorbate is reduced by half to be \( m = 99 \) while \( \epsilon \) (the interaction strength between the adsorbate and graphene) remains unchanged. In the third system \((m, \epsilon/2)\), the mass of the adsorbate is unchanged at \( m = 197 \) while the interaction strength \( \epsilon \) between the adsorbate and the graphene is reduced by half.

In all three systems, there is a sharp decrease in the \( Q \)-factor at different values of the actuation energy parameter \( \alpha \), which results from the diffusion of the adsorbate. These results also follow physical intuition. For example, diffusion happens at smaller \( \alpha \) for adsorbates that have a weaker bonding strength with graphene. In contrast, diffusion happens at larger \( \alpha \) for adsorbates with less mass, since these atoms have a smaller kinetic energy at the same temperature as larger mass adsorbates, and therefore more energy via the nonlinear actuation parameter \( \alpha \) is needed to induce diffusion for the smaller mass atoms. These results may serve as useful guidelines for experimentalists to verify our theoretical predictions, as different types of adsorbate (of different mass and/or different interaction strength with graphene) are usually observed experimentally.

![Figure 2](image-url)

**Figure 2.** The resonant frequency versus actuation energy parameter \( \alpha \). (a) The resonant frequency for a pure GNMR (circular points) and a GNMR with one Au atom adsorbed (pentagonal points). (b) The magnitude of frequency shift induced by the adsorption of a single Au atom for different actuation energy parameters \( \alpha \).

![Figure 3](image-url)

**Figure 3.** The mean free path of the Au atom as a function of the actuation energy parameter \( \alpha \). A sharp jump in the mean free path occurs at \( \alpha_c = 2.0875 \pm 0.0125 \), implying diffusion of the Au atom for \( \alpha > \alpha_c \). The left top inset shows smooth resonant curves for smaller actuation energy parameters, \( \alpha = 0.1 \) and 1.5. The right bottom inset shows significant diffusion-induced noise in the resonant curves for large actuation energy parameters, \( \alpha = 4.0 \) and 10.0.
Figure 4. The trajectory history of the adsorbed Au atom for actuation energies of $\alpha = 1.0, 2.075, 2.1$ and $4.0$ from left to right. The first two panels show localized vibration of the Au atom, while the last two panels clearly show diffusion of the Au atom.

Figure 5. The effect of the diffusion of the Au atom on the $Q$-factor of the GNMR. (a) The $Q$-factor of a Au adsorbed GNMR remains extremely high for $\alpha < \alpha_c$, but becomes three orders of magnitude smaller for $\alpha$ greater than the critical value $\alpha_c$. The energy decay in (c) is caused by the diffusion of the adsorbed Au atom shown in figure 4.

4. Discussion

The above discussion has established that the mass sensitivity of the GNMR can be enhanced by driving the mechanical oscillations into the nonlinear regime using a larger actuation energy. The remainder of this paper will be devoted to explaining the mechanism that enables the nonlinear-induced resonant frequency, and thus mass sensitivity enhancement.

As illustrated in figure 7, the GNMR is initially flat (gray points on the horizontal line) with length $L$. After exciting the mechanical oscillations via the actuation energy $\Delta E = \alpha \epsilon_k^0$, the GNMR oscillates with amplitude $A = \sqrt{\Delta E / m \omega^2}$, where $\omega = 2\pi f$ is the angular frequency of the mechanical oscillation and $m$ is the total mass of the system. In the derivation of the oscillation amplitude, the thermal vibrations at 4.2 K have been ignored due to the very low temperature conditions.
GNMR is obtained analogously. As previously discussed, the strain and the actuation energy that is applied to the GNMR, we get a concise relationship between the effective strain and the actuation energy of the GNMR from the MD simulations. We focus on the difference between the effective strain for the oscillating GNMR and the original, flat GNMR (gray points on the horizontal line).

This effective strain is denoted by $\epsilon_{\text{f}}$. Similarly, the shape of the oscillating GNMR is under an 'effective strain' as compared to the original, flat GNMR (red dots) in figure 7, our key insight is that the oscillating regime is actually due to the effective strain that is induced in the oscillating GNMR.

In figure 8, we show that the effective strain is indeed responsible for the enhancement of the resonant frequency with increasing $\alpha$. We present the resonant frequency for the GNMR in (a) and the GNMR with adsorption of a single Au atom in (b). The circles represent the resonant frequency of the GNMR due to different actuation energies $\alpha$, i.e. $f(\epsilon_\alpha)$, while the dashed lines (blue online) are the resonant frequencies $f(\epsilon_\alpha)$ for the GNMR that was pre-stretched with a tensile strain $\epsilon_\alpha$ before actuation, where the strain $\epsilon_\alpha$ for actuation energy $\alpha$ is calculated by equation (2).

Figures 8(a) and (b) both show good agreement between the resonant frequencies obtained by increasing the actuation energy $\alpha$ and by applying a tensile strain $\epsilon_\alpha$. This agreement verifies that the effective strain is the cause of the increase of the resonant frequency that we have previously observed by increasing $\alpha$. The agreement is particularly good for $\alpha < 1.0$, though there is an increasing discrepancy for larger $\alpha$. The solid lines (red online) show that this discrepancy is greatly improved by considering a high-order correction in the square of the oscillation amplitude $A^2$. This high-order correction...
is demonstrated in figure 8(c), where \( u_z \) on the y-axis of figure 8(c) is the oscillation of the midpoint of the GNMR obtained from the MD simulation, and where figure 8(c) is plotted using a log–log scale. \( \langle u_z^2 \rangle \) deviates from linear behavior because of the phonon–phonon scattering. Although the temperature is quite low (4.2 K), the phonon–phonon scattering is still important for the first bending mode (i.e. the mode of mechanical oscillation) because this mode is in a highly non-equilibrium state. In other words, because the first bending mode has been mechanically acted with very large amplitude, it is driven far away from its thermal equilibrium state at 4.2 K. Our MD simulation gives \( \langle u_z^2 \rangle \propto \alpha^{0.79} \) for the pure GNMR. However, we have assumed \( \langle u_z^2 \rangle \propto \alpha^{1.0} \) in our derivation of the effective strain. The effective strain \( \epsilon_{\alpha} \propto \langle u_z^2 \rangle \), which is why we have \( \epsilon_{\alpha} \propto \alpha \). Therefore, the effective strain must be modified by \( \epsilon'_{\alpha} = \epsilon_{\alpha}/\alpha^{0.21} \), where \( \epsilon_{\alpha} \) is the effective strain without correction. A similar correction factor of \( \alpha^{0.28} \) can be made for the effective strain of the GNMR with a single adsorbed Au atom.

We note that figure 8(c) shows a small oscillation amplitude (0.8 Å) for the system simulated in this work, due to its small size. The amplitude \( U \) actually depends on the length \( L \) of the graphene as \( U = 2.43 \times \sqrt{\omega L} \) Å, where the frequency \( \omega \) is length dependent. For the system we have simulated, this formula gives \( U \approx 0.8 \) Å at \( \alpha = 1.0 \), which is in good agreement with the MD simulation results in figure 8(c). Considering the flexural property of the bending mode in graphene, i.e. \( \omega \propto 1/L^2 \), we can obtain the oscillation amplitude of an arbitrary system length as \( U = U_0(L/L_0)^{2} \), where \( L_0 = 34 \) Å and \( U_0 = 0.8 \) Å for our simulated system at \( \alpha = 1.0 \). For instance, the mechanical oscillation amplitude becomes 67 nm if the system length is 100 nm. If the system length is 1 μm, then the mechanical oscillation amplitude is very large, i.e. 6.7 μm. These results show that \( \alpha = 1 \) is large enough to induce oscillations that can be clearly distinguished from random thermal fluctuations in experiments.

Previously, we have established that the effective strain mechanism can explain the enhancement of the resonant frequency that occurs by increasing \( \alpha \). However, we also needed to perform a high-order correction for large actuation energies to account for the phonon–phonon scattering that is still present even at the low (4.2 K) temperature due to the highly non-equilibrium behavior resulting from the large actuation energy. However, if the temperature is reduced to 0 K there is no phonon–phonon scattering for the first bending mode because this mode cannot decay without the assistance of other vibrations according to symmetry selection rules [28]. As a result, at 0 K, the enhancement of the resonant frequency by applying a tensile strain before actuation should agree with that obtained by increasing the actuation energy, without any high-order correction.

Indeed, we observe this result in figure 8(d). We emphasize that for these two simulations of the GNMR, the GNMR exists initially in an energy minimized configuration at 0 K, and the applied actuation energy is \( \Delta E = \alpha E_k^0 \). Because the kinetic energy \( E_k^0 \) is 0 at 0 K, we use \( E_k^0 = E_k(4.2 \) K), where \( E_k(4.2 \) K) is the total kinetic energy at 4.2 K, for both of the two simulations shown in figure 8 (d).

The solid line is the kinetic energy for the GNMR that is actuated by the large actuation energy parameter \( \alpha = 2.0 \). The dashed line (blue online) is the kinetic energy of the GNMR that is pre-stretched in tension by 0.4% before actuation, where 0.4% is chosen as it is also the effective strain that results from an actuation energy of \( \alpha = 2.0 \). The pre-stretched GNMR is actuated with a very small actuation energy of \( \alpha = 0.001 \) so that the effect from \( \alpha \) can be ignored and only strain is important in this case. We observe exactly the same resonant frequency in the mechanical oscillation in these two very different situations. By Fourier transformation of the time histories of these two kinetic energies, we get the same resonant frequency of 270 GHz, which further establishes the equivalent effect of applied tensile strain and different actuation energies on the resonant frequencies of GNMRs.

5. Conclusion

In conclusion, we used classical MD simulations to demonstrate that a significant enhancement of the mass sensitivity of graphene nanomechanical resonators can be achieved by driving them into the nonlinear oscillation regime. The enabling mechanism was determined to be the effective strain that is induced in the nanoresonators due to the nonlinear oscillations. A simple analytic expression relating the effective strain to the actuation energy was obtained, and shown to be quite accurate for moderate actuation energies. The key implication is that it should be possible for experimentalists to directly incorporate the present findings to enhance the sensitivity of graphene-based mass sensors simply by actuating the graphene nanomechanical resonators into the nonlinear oscillation regime.

Acknowledgments

This work is supported by the Grant Research Foundation (DFG) of Germany. HSP acknowledges support from the Mechanical Engineering Department of Boston University.

References

[1] Ekinci K L and Roukes M L 2005 Rev. Sci. Instrum. 76 061101
[2] Arlett J L, Myers E B and Roukes M L 2011 Nature Nanotechnol. 6 203
[3] Eom K, Park H S, Yoon D S and Kwon T 2011 Phys. Rep. 503 115
[4] Barton R A, Parpia J and Craighead H G 2011 J. Vac. Sci. Technol. B 29 050801
[5] Jensen K, Kim K and Zettl A 2008 Nature Nanotechnol. 3 533
[6] Lassagne B, Garcia-Sanchez D, Aguasca A and Bachhold A 2008 Nano Lett. 8 3735
[7] Chiu H-Y, Hung P, Postma H W Ch and Bockrath M 2008 Nano Lett. 8 4342
[8] Lee C G, Wei X D, Kysar J W and Hone J 2008 Science 321 385
[9] Jiang J-W, Wang J-S and Li B 2009 Phys. Rev. B 80 113405
[10] Jiang J-W, Wang J-S and Li B 2010 Phys. Rev. B 81 073405
[11] Bunch J S, van der Zande A M, Verbridge S S, Frank I W, Tanenbaum D M, Parpia J M, Craighead H G and McEuen P L 2007 Science 315 490
[12] van der Zande A M, Barton R A, Alden J S, Ruiz-Vargas C S, Whitney W S, Pham P H Q, Park J, Parpia J M, Craighead H G and McEuen P L 2010 Nano Lett. 10 4869
[13] Chen C, Rosenblatt S, Bolotin K I, Kalb W, Kim P, Kymissis I, Stormer H L, Heinz T F and Hone J 2009 Nature Nanotechnol. 4 861
[14] Eichler A, Moser J, Chaste J, Zdrojek M, Wilson-Rae I and Bachtold A 2011 Nature Nanotechnol. 6 339
[15] Kim S Y and Park H S 2009 Nano Lett. 9 969
[16] Jiang J-W and Wang J-S 2012 J. Appl. Phys. 111 054314
[17] Buks E and Yurke B 2006 Phys. Rev. B 74 046619
[18] Dai M D, Eom K and Kim C-W 2009 Appl. Phys. Lett. 95 203104
[19] Atalaya J, Kinaret J M and Isacsson A 2010 Europhys. Lett. 91 48001
[20] Dai M C, Kim C-W and Eom K 2012 Nanoscale Res. Lett. 7 499
[21] Kim S Y and Park H S 2008 Phys. Rev. Lett. 101 215502
[22] Kim S Y and Park H S 2010 Nanotechnology 21 105710
[23] Kim S Y and Park H S 2009 Appl. Phys. Lett. 94 101918
[24] Qi Z and Park H S 2012 Nanoscale 4 3460
[25] Brenner D W, Shenderova O A, Harrison J A, Stuart S J, Ni B and Sinnott S B 2002 J. Phys.: Condens. Matter 14 783
[26] Noé S 1984 J. Chem. Phys. 81 511
[27] Hoover W G 1985 Phys. Rev. A 31 1695
[28] Born M and Huang K 1954 Dynamical Theory of Crystal Lattices (Oxford: Oxford University Press)