Elasticity in strongly interacting soft solids: polyelectrolyte network

J. Wilder* and T.A. Vilgis†
Max-Planck-Institut für Polymerforschung, Postfach 3148, 55021 Mainz, Germany

This paper discusses the elastic behavior of a very long crosslinked polyelectrolyte chain (Debye-Hückel chain), which is weakly charged. Therefore the response of the crosslinked chain (network) on an external constant force \( f \) acting on the ends of the chain is considered. A selfconsistent variational computation of an effective field theory is employed. It is shown, that the modulus of the polyelectrolyte network has two parts: the first term represents the usual entropy elasticity of connected flexible chains and the second term takes into account the electrostatic interaction of the monomers. It is proportional to the squared crosslink density and the Debye-screening parameter.

PACS: 05.20.-y, 36.20.-r, 61.41.+e

I. INTRODUCTION

Polyelectrolytes are of fundamental importance in a wide range of academic sciences to applications. It ranges from life sciences such as biology or biochemistry to industrial and practical applications in daily life products. A typical example for the latter are superabsorber materials. These consist of highly crosslinked polyelectrolyte networks which are strongly interacting elastic materials.

The theoretical interest in polyelectrolytes reaches back to the early days of polymer science (see e.g. [1]). Nevertheless they belong to the least understood systems in macromolecular science [2], since there are difficulties to apply renormalization group theories and scaling ideas in which long ranged (i.e. Coulomb) forces are present. Only very recently novel types of field theoretic attempts brought progress [3].

In the present paper we aim for a theory of the elasticity of polyelectrolyte networks. This is a non trivial task, since most of the classical and modern theories neglect the effect of interactions on elasticity. In neutral networks the interactions are mainly given by excluded volume forces. In the dry network state these can be safely neglected for most cases, since in such dense systems like polymer melts excluded volume interactions are screened largely [4,5]. However, if solvent is added to the network and the network starts to swell, problems arises. Early theories by Flory [6] suggested that the elastic part of the free energy and the solvent part, i.e., a Flory-Huggins type term, can be added. Later on this concept has been named the Flory-Rehner-Hypothesis [7] in the context of swelling experiments. The network state would be determined by the minimum of the total free energy. Indeed such approximations are used in a wide range of application for rubberlike materials in the swollen state [8]. The comparison with experiments seem to be reasonable, i.e., in all cases the modulus is found to be proportional to the crosslink density, although from a theoretical point of view the simple addition of the two parts of the free energy must be wrong [9]. It must be wrong, because a complete formulation of the partition function suggests immediately the appearance of cross terms. In a subsequent paper [10] we will analyse the Flory-Rehner-Hypothesis in the context of thermodynamics and its application. There it is shown that corresponds to an approximation which neglect fluctuations completely. In the present paper this is not the central point and we restrict ourselves to compute the elastic response of a polyelectrolyte network including fluctuations on a variational level.

In neutral networks, however the approximations seem to be not too bad, because the interactions are relatively short ranged and weak. Moreover the equilibrium swelling degree is then given by the \( c^* \)-network. When polyelectrolyte networks are considered, we cannot expect that the Flory approximation holds. The interactions are long ranged and very strong compared to excluded volume interactions. Here a strong interplay of elastic degrees of freedom and interactions must be expected. The reason is very simple: The strong interactions change the physical nature and the conformation of a charged polymer chain strongly compared to the neutral chain. The state at rest, i.e., a network strand without application of an external force does not contain as many degrees of conformation as the equivalent neutral chain. Its conformation ranges, depending on ionization and salt content of the solution, from excluded volume behavior to a rod like behavior. Thus a more detailed theory is needed to compute the elastic modulus of charged

*e-mail: wilder@mpip-mainz.mpg.de
†e-mail: vilgis@mpip-mainz.mpg.de
and highly interacting gels. Nevertheless the Flory approximation has been employed also for strongly interacting polyelectrolyte gels in bad solvent to study the phase diagram \[1\]. Moreover also recent suggestions \[12\] have claimed an unchanged modulus for polyelectrolyte networks. We will see later, however, the reasons for these statements.

Most of the classical network theories rely on "single chain models". This is to say that the elasticity of the network can be roughly computed by studying first the elasticity of a single chain. The elastic properties of the entire network is then supposed to be given by the partition function of the single chain raised to the power of the number of chains. Such ways of computations hold strongly only for weakly interacting systems. Again in neutral networks the interactions, i.e., the excluded volume forces, are weak since they are screened, but in polyelectrolyte networks these type of assumptions do not seem to be reasonable. Another drawback of these single chain type theories is that they ignore the effect of quenched disorder stemming from the crosslinks completely. Although some progress for polyelectrolyte networks has been achieved by single chain theories \[13\], which are based on blob pictures \[15\], it is necessary to study a network model including the quenched disorder produced by the crosslinks (see e.g. \[1\] for a review). It has to be kept in mind that so far only the limits for large and small screening parameters are studied. Large screening parameters (in terms of a Debye - Hückel approximation) correspond to a large salt concentration. This regime is relatively uninteresting, since it corresponds to the good solvent regime. For small screening the chains are significantly stretched, and the chains are in the polyelectrolyte regime \[16\]. For networks, however, these two limits alone are not sufficient. In strongly crosslinked networks the chain pieces between two crosslinks can easily be of the same order as the Debye - Hückel screening length. Therefore a more elaborated analysis must be carried out.

In a previous paper \[16\] we had investigated the coupling between elasticity and interactions in a single chain. This model calculation was carried out to show that a strong coupling between elastic (conformational) degrees of freedom and electrostatic interactions exists. Of course, unlike as in neutral network systems it is not sufficient to consider the elasticity of a single chain and generalize the results to a corresponding network of a large number of such chains. In the case of polyelectrolyte networks the chains are strongly interacting with each other. Therefore any assumption on weakly interacting chains fails. Nevertheless the computational method used in \[16\] has shown to be useful. The results presented there ended up in a two regime blob picture. For small forces the de Gennes \[1\] electrostatic blob model was recovered. In this regime the chain was relatively easy to deform. For larger forces a change in the elasticity was predicted. Then the chain entered in a Pincus regime of the prestretched chain. The results have been confirmed by simulation and by a self consistent variational principle. Indeed the method developed in \[16\] is appropriate to the intermediate regime between strong and weak screening.

In this paper we extend our considerations concerning the single polyelectrolyte chain \[16\] to a polyelectrolyte network. The simplest version of a polymer network was introduced by Dean and Edwards \[17\], which consists of a very (macroscopically) long crosslinked chain. In our case there is no difference between a network made of many chains or one long chain, since we assume to be deep in the solid phase, i.e., well beyond the vulcanization threshold. As in the case of a single polyelectrolyte chain \[16\] we are interested in the force-size relationship of the network in a good solvent. In contrast to the classical theories the effects of interactions are now taken into account explicitly. For simplicity we assume a Debye-Hückel potential for the electrostatic interaction. It is of the form \( V(r) \propto 1/r \exp(-r/\lambda) \). This might not be the best choice to reproduce recent simulation data \[18\], which had shown that the Debye-Hückel potential is not always a good approximation, but the advantage of the potential is that its range is controlled by a single parameter, i.e., the Debye screening length \( \lambda \). Indeed little is known about the interplay of elasticity (conformation) and strength of the interactions in the theory of network elasticity. The aim of the present paper is to learn something about this interplay. Therefore we apply an external force on the ends of the chain. To do so, we employ a variational principle and determine the effective propagator of the chain, which allows statements about the interplay of conformation and interactions on the elasticity. In the following we will neglect the effect of entanglements completely, since we are mainly interested in the contributions of the interactions to elasticity. The results given below will then mainly apply for unentangled networks. To produce simple estimates of entanglement contributions tube like models can be employed \[14\]. We expect, however, that the effects of entanglements will not be very different from those in neutral networks, as long as their number is given. The main problem would then be to compute the mean number of entanglements by the presence of the electrostatic interactions.

The starting point for the computation of the elastic free energy is the Green function of the crosslinked chain without an external force, similar to our calculations of the single polyelectrolyte chain \[16\]. The force is treated through the analytic continuation of the Fourier transformed Greens function to the complex plane. After having introduced a field theory the problem is mapped on a Gaussian field theory with a propagator that formally in the Fourier space can be written down exactly by making use of the proper self energy. According to the well known Feynman variational inequality the sum of the Gaussian free energy and the mean-value of the interacting potential has to be minimized with respect to the proper self energy, which is our variational parameter. This leads to a non-linear integral equation for the proper self energy, which can be solved approximately.

The paper is organized as follows. In the next section we present the underlying model and introduce the crosslinks. In section III we formulate a field theory and calculate the variational equation for the proper self energy. In section
IV this variational equation is solved approximately. These sections will be written out in more detail, since this is - to our knowledge - the first time that a strongly interacting network is investigated by this technique. Thus the mathematically interested reader might find the main steps of the computation. The results are presented in section V. The paper ends with the discussion of the results.

II. MODEL AND THEORETICAL BACKGROUND

The starting point for the field theoretic computation is a network formed out of a macroscopically long chain by the instantaneous introduction of a sufficiently large number of crosslinks in the liquid phase (see for example [17,19]). We restrict ourselves to a network, which consists of flexible, weakly charged strands. Consequently the Edwards model is an appropriate tool to describe the network [20]. Therefore let us choose as Hamiltonian for the charged chain in aqueous solution:

\[ \beta H_E[r; f] = \frac{3}{2l^2} \int_0^{N_{tot}} ds \left( \frac{dr}{ds} \right)^2 + \beta \int_0^{N_{tot}} ds \int_0^{N_{tot}} ds' \exp \left\{ -\kappa |r(s) - r(s')| \right\} \],

where \( r(s) \) represents the chain conformation in three dimensions as a function of the contour variable \( s \), \( b = e^2/4\pi\epsilon_0\kappa_B T \) is the Bjerrum-length, \( \beta = (k_B T)^{-1} \), where \( k_B \) is the Boltzmann constant and \( T \) denotes the absolute temperature. \( l \) is the Kuhn segment length, \( z \) is the monomer charge in units of \( e \), \( \epsilon_0 \) is the dielectric constant and \( \epsilon_r \) the relative dielectric constant. \( N_{tot} \) stands for the bare number of monomers on the chain, \( f \) is the external force and \( \kappa^{-1} \) denotes the Debye-Hückel screening length \( \lambda \). For the introduction of crosslinks we choose the standard way suggested by Dean and Edwards [17]. We assume for mathematical convenience four functional crosslinks which join two arbitrary segments \( r(s_i) \) and \( r(s_j) \) along the chain. Of course, the value for the free energy then depends on the specific choice of the pairs of monomers, but on macroscopic scale only the statistical average on any crosslink configuration is of importance. Nevertheless this requires non-Gibbsian statistical mechanics in the sense that the crosslink positions represent quenched degrees of freedom.

The basic problem for the determination of the free energy of the network is the presence of quenched disorder, which is contained in the permanent crosslinks. The formation of a crosslink, i.e., the linkage between two arbitrary monomers \( r(s_i) \) and \( r(s_j) \) represents quenched disorder, since the segments are joined together for all times and for all thermodynamic situations. Of course, the experimental relevant free energy \( F \) depends on the crosslink configuration and crosslink realization \( S \). The actual crosslink configuration \( S \) is not known in detail, thus the technical difficulty is to average the free energy \( F(S) \) over all possible crosslink realizations. To do so, it is generally assumed that the corresponding distribution \( P(S) \) can be determined.

Let us perform the outline of the idea in more detail. To calculate the free energy \( F \) of the network, we have to take the statistical average over all crosslink configurations \( S \). This represents the fact, that \( F \) is a self averaging quantity.

\[ F(N_{tot}, N_c) = -k_B T \int dS \mathcal{P}(S) \ln Z(S). \]  

\( Z(S) \) is the constrained partition function for a network with the crosslink configuration \( S \), \( N_c \) is the number of crosslinks and \( \mathcal{P}(S) \) is the crosslink distribution function. Since we assume that the crosslinks are instantaneously introduced in the liquid phase, \( \mathcal{P}(S) \) is yielded by the constrained partition function of the liquid phase, which is defined in terms of a path integral as:

\[ Z^{(0)}(S) = \int Dr(s) \exp(-\beta H_E) \prod_{(i,j)} \delta[r(s_i) - r(s_j)], \]

where \( (i,j) \) denotes that the \( i \)-th and the \( j \)-th monomer are close to each other in the liquid phase, which means that they can form one of the \( N_c \) crosslinks. Consequently the crosslink distribution function \( \mathcal{P}(S) \) is simply:

\[ \mathcal{P}(S) = \frac{Z^{(0)}(S)}{\int dS' Z^{(0)}(S')} \].

In the following it appears to be reasonable to assume that the so chosen distribution function does not depend on the specific deformation of the network. Note that \( Z(S) \) differs generally from \( Z^{(0)}(S) \). Since we are interested in deformations of the network, \( Z(S) \) is the partition function of the deformed network.

3
To calculate the free energy $F$ (Eq. (2)) explicitly it is convenient to make use of the so-called replica trick \cite{17}. This, so far, purely mathematical trick relies on the identity

\[
\ln z = \left. \frac{\partial z^m}{\partial m} \right|_{m=0}
\]

Define

\[
F_m(N_{\text{tot}}, N_c) = -k_B T \ln \int \mathcal{D} S \, Z^{(0)}(S) Z^m(S),
\]

where $m$ is the replica index. Eq. (5) shows the origin of the technical term "replica method". By the use of the mathematical trick $m$ copies of the system are produced. The free energy $F$, which is averaged over the disorder of the crosslinks, reads \cite{17}:

\[
F(N_{\text{tot}}, N_c) = \left. \frac{\partial F_m(N_{\text{tot}}, N_c)}{\partial m} \right|_{m=0}
\]

As in the previous paper, the free energy $F$ is calculated by making use of its relation to the corresponding distribution functions and Green functions of the corresponding propagator (see \cite{16} for the technical details).

III. FIELD-THEORETICAL FORMULATION

In the following we will give an outline of the computation of the network elasticity, i.e., our main aim is to compute the low deformation modulus of the polyelectrolyte network. Therefore we start from a concentrated polyelectrolyte solution, consisting of one macroscopic chain and the appropriate number of counterions to satisfy the condition of electro-neutrality. Then the crosslinks are introduced instantaneously by the process described above. The following chapter will be very formal, but we think that it is important to do so, since it turned out that none of the methods employed for neutral networks can be used in the present context. The main reason for this is that here we do not have the option of formulating the problem in terms of one length scale, i.e., the meshsize, but must take into account the range of the interaction.

For the relevant quantities to compute it is necessary to consider the correlation function $G(\mathbf{r}, N_{\text{tot}}, N_c, \mathbf{f}, S)$ of a crosslinked polyelectrolyte chain in replica space, where $\mathbf{f} = (r_0, r_1, \ldots, r_m)$ is the replicated $(m+1)$-dimensional end-to-end vector of the chain and $S$ the specific crosslink configuration:

\[
G(\mathbf{r}, N_{\text{tot}}, N_c, \mathbf{f}, S) = \int_{r_0(0)=0}^{r_0(N_{\text{tot}})=0} \mathcal{D} r_0(s) \int_{r_1(0)=0}^{r_1(N_{\text{tot}})=r_1} \mathcal{D} r_1(s) \ldots \int_{r_m(0)=0}^{r_m(N_{\text{tot}})=r_m} \mathcal{D} r_m(s)
\]

\[
\times \exp(-\beta \hat{H}_{E}[\mathbf{r}, \mathbf{f}]) \prod_{p=0}^{m} \prod_{(i,j)} \delta[r_p(s_i) - r_p(s_j)]
\]

with $p$ the replica index. The force $\mathbf{f}$ is only acting on the ends of the final states (replica index 1 to $m$) of the chain. Thus the replicated Hamiltonian $\hat{H}_{E}[\mathbf{r}, \mathbf{f}]$ reads

\[
\beta \hat{H}_{E}[\mathbf{r}, \mathbf{f}] = \frac{3}{2 l^2} \sum_{p=0}^{m} \int_{0}^{N_{\text{tot}}} ds \left( \frac{dr_p}{ds} \right)^2 + \beta \sum_{p=0}^{m} \int_{0}^{N_{\text{tot}}} ds \int_{0}^{s} ds' \exp \left\{ -\kappa \frac{|r_p(s) - r_p(s')|}{|r_p(s) - r_p(s')|} \right\}
\]

The important observation is that the replicated Hamiltonian $\hat{H}_{E}$ separates in the different replicas. The coupling of the replicas comes into play when the average over the distribution $\mathcal{P}(S)$ is performed. If we use the standard distribution \cite{17} the Green function must be computed upon the effective Hamiltonian

\[
\hat{H} = \hat{H}_{E} - z_c \int_{0}^{N_{\text{tot}}} \int_{0}^{N_{\text{tot}}} ds \sum_{p=0}^{m} \delta(r_p(s) - r_p(s'))
\]
The latter equation shows the difficulty of the problem, i.e., all replicas are coupled. Below we choose a different way as suggested by Edwards [18] and Panyukov [19]. We must employ field theoretic methods (as also done in [19]), but the treatment of the field theory is very different, since the symmetry of the problem is not of the same nature as in the case of neutral networks.

It is easy to show that the Greens function in the Fourier transformed replica-space depending on a constant force $f$ can be calculated by a zero force Greens function. The force can be reintroduced by the analytic continuation of the Fourier-space to the complex plane, which means in detail:

$$G(\hat{k}, N_{\text{tot}}, N_c, f, S) = \int d\omega \exp\{-i(\hat{k} - i\beta\hat{f})\omega\}G(\hat{k}, N_{\text{tot}}, N_c, f = 0, S) = G(k^{(0)}(N_{\text{tot}}, N_c, f = 0, S))$$

Here $\hat{f}$ is the 3$(m+1)$-dimensional force vector $(0, f, \ldots, f)$. This is exactly the same mathematical procedure as we had already used in the previous calculation concerning the single polyelectrolyte chain (see [16]). As a consequence of Eq. (10) in the following we neglect the force term in the Hamiltonian and first calculate a zero force correlation function.

The grand canonical correlation function $\hat{G}(\hat{k}, \mu_0, z_c, f)$ in replicated Fourier space, where $\hat{k}$ is the wave vector in the $3(m+1)$-dimensional Fourier transformed replica space, $\mu_0$ is the chemical potential of the monomers and $z_c$ is the fugacity of the crosslinks, can be calculated by the introduction of de Gennes’ zero-component field theory (see for example [19])

$$\hat{G}(\hat{k}, \mu_0, z_c, f) = \lim_{n \to 0} \int D\psi \psi_1(\hat{k})\psi_1(-\hat{k}) \exp\{-\beta H[\psi]\}.$$  

$H[\psi]$ in Eq. (11) is the zero force field theoretical Hamiltonian expressed by the $n$-component field $\psi$, which in Fourier space reads [10,19]:

$$H[\psi(q)] = \int_q \left[ \frac{\mu}{2} \psi^2(q) + \frac{l^2}{2} \psi^2(q) \psi(-q) \right]$$

$$- \frac{z_c}{8} \int_{q_1,q_2,q_3,q_4} \psi(q_1)\psi(q_2)\psi(q_3)\psi(q_4)\delta(q_1 + q_2 + q_3 + q_4)$$

$$+ \sum_{k=0}^{m} \left[ \int_{q_1,q_2} \psi(q_1)\psi(q_2) \prod_{l \neq k} \delta(q_1^{(l)} + q_2^{(l)}) \right]$$

$$\times \left[ \int_{q_3,q_4} \psi(q_3)\psi(q_4) \prod_{l \neq k} \delta(q_3^{(l)} + q_4^{(l)}) \right] V^{(k)}(q_3^{(k)} + q_4^{(k)})\delta(q_1^{(k)} + q_2^{(k)} + q_3^{(k)} + q_4^{(k)})$$

where $V^{(k)}(q)$ is the Fourier transform of the Debye-Hückel potential in the $k$-th replica, $\psi$ is a $n$-component vector field and $\int$ is an abbreviation for the integral notation $\int dq/(2\pi)^d$ with $d$ the dimension of the vector $q$. In Fourier space $\hat{G}(\hat{k}, \mu_0, z_c)$ can be written exactly as:

$$\hat{G}(\hat{k}, \mu_0, z_c) = \left( \mu_0 + \frac{l^2}{6} \hat{k}^2 + \Sigma(\hat{k}, z_c) \right)^{-1}$$

where $\Sigma(\hat{k}, z_c)$ denotes the proper self energy in replica space.

Since we do not know the exact proper self energy $\Sigma(\hat{k}, z_c)$, we have to calculate it approximately. Therefore we now consider an approximate correlation function $\hat{G}(\hat{k}, \mu_0, z_c)$ with an approximate proper self-energy $M(\hat{k}, z_c)$:

$$\hat{G}(\hat{k}, \mu_0, z_c) = \left( \mu_0 + \frac{l^2}{6} \hat{k}^2 + M(\hat{k}, z_c) \right)^{-1}$$

To proceed with a variational principle we define the Gaussian Hamiltonian $\mathcal{H}$ by

$$\beta \mathcal{H}[\psi] = \frac{1}{2} \int \hat{\psi}(\hat{k})\hat{G}^{-1}(\hat{k}, \mu_0, z_c)\hat{\psi}(\hat{k})$$
The correlation function $\tilde{G}(\vec{k}, \mu, z_c)$ can be calculated within the zero-component field theory of de Gennes (see for example [10]):

$$\tilde{G}(\vec{k}, \mu, z_c) = \lim_{n \to 0} \int \mathcal{D}\bar{\psi} \psi_1(\hat{k}) \psi_1(-\hat{k}) \exp\{-\beta \mathcal{H}[\bar{\psi}]\}$$  \hspace{1cm} (16)

In this notation the well-known Feynman inequality, which can be taken for the calculation of the approximate proper self energy $M(\vec{k}, z_c)$, is given by:

$$F \leq F + \langle H - \mathcal{H} \rangle_{\mathcal{H}}$$  \hspace{1cm} (17)

where

$$\langle \ldots \rangle_{\mathcal{H}} = \lim_{n \to 0} \frac{\int \mathcal{D}\bar{\psi} \ldots \exp\{-\beta \mathcal{H}\}}{\int \mathcal{D}\bar{\psi} \exp\{-\beta \mathcal{H}\}}$$  \hspace{1cm} (18)

is the mean-value and $F$ the free energy with respect to $\mathcal{H}$. The right hand side of the inequality ([10]) has to be minimized with respect to $M$. $F$ and $(H - \mathcal{H})_{\mathcal{H}}$ can be written in terms of the correlation function $\tilde{G}(\vec{k}, \mu, z_c)$:

$$\beta F = -\frac{n}{2} V \int_{\mathcal{q}} \ln \left[ \tilde{G}(\mathbf{q}, \mu, z_c) \right]$$  \hspace{1cm} (19)

where $V$ is the volume of the replica space. As can be shown easily the second term of the right hand side of inequality ([17]) is [21]

$$\beta < H - \mathcal{H} >_{\mathcal{H}} = -\frac{n}{2} V \int_{\mathcal{q}} m(M(\mathbf{q}, z_c)\tilde{G}(\mathbf{q}, \mu, z_c)) + \frac{\pi b z^2 n^2 (m+1) V_0^m V}{2 \kappa^2} \left( \int_{\mathcal{q}} \tilde{G}(\mathbf{q}, \mu, z_c) \right)^2$$

$$+ \frac{\pi b z^2 n (m+1) V}{2} \int_{\mathcal{q}} \frac{\tilde{G}(\mathbf{q}, \mu, z_c)^2}{(\mathbf{q}^{(0)} + \mathbf{q}^{(0)} + \mathbf{q}^{(0)})^2 + \kappa^2} \prod_{i=1}^{m} \delta(\mathbf{q}^{(i)} + \mathbf{q}^{(i)})$$

$$- \frac{z_c}{2} (n^2 + 2n) V \left( \int_{\mathcal{q}} \tilde{G}(\mathbf{q}, \mu, z_c) \right)^2$$  \hspace{1cm} (20)

with $V_0$ the volume of a single replica segment and $V$ the volume of the whole replica-space. As we want to determine the approximate proper self energy $M(\vec{k}, z_c)$, this function should be the variational parameter. Consequently the general minimization condition reads:

$$\frac{\delta}{\delta M(\mathbf{q}, z_c)} (F + \langle H - \mathcal{H} \rangle_{\mathcal{H}}) = 0$$  \hspace{1cm} (21)

where $\delta/\delta M(\mathbf{q}, z_c)$ denotes the functional derivative with respect to $M(\mathbf{q}, z_c)$. After inserting Eqs. (18) and (20) into Eq. (21) one obtains

$$M(\vec{k}, z_c) = \frac{2 \pi b z^2 n (m+1) V_0^m}{\kappa^2} \int_{\mathcal{q}} \frac{1}{\mu + \frac{\rho^2}{2} \hat{q}^2 + M(\mathbf{q}, z_c)}$$

$$+ 4 \pi b z^2 \int_{\mathcal{q}^{(0)}} \frac{1}{(\mathbf{q}^{(0)} + \mathbf{k}^{(0)})^2 + \kappa^2 \mu + \frac{\rho^2}{6} \left( \mathbf{q}^{(0)} , \mathbf{k}^{(0)} (1, \ldots , m) \right)^2 + M(\mathbf{q}^{(0)} , \mathbf{k}^{(0)} (1, \ldots , m))} \frac{1}{2}$$

$$+ 4 \pi b z^2 m \int_{\mathcal{q}^{(i)}} \frac{1}{(\mathbf{q}^{(i)} + \mathbf{k}^{(i)})^2 + \kappa^2 \mu + \frac{\rho^2}{6} \left( \mathbf{q}^{(i)} , \mathbf{k}^{(i)} (1, \ldots , m) \right)^2 + M(\mathbf{k}^{(0)} , \mathbf{q}^{(i)} (1, \ldots , m))} \frac{1}{2}$$

$$+ \frac{z_c}{2} (n+2) \int_{\mathcal{q}} \frac{1}{\mu + \frac{\rho^2}{2} \hat{q}^2 + M(\mathbf{q}, z_c)}$$  \hspace{1cm} (22)

where the short hand notation $\mathbf{k}^{(i, \ldots , m)} = (\mathbf{k}^{(i)}, \ldots , \mathbf{k}^{(m)})$ is introduced. This is a non-linear integral equation for $M(\mathbf{q}, z_c)$, which in the following has to be solved approximately, since the exact solution is unknown.
IV. APPROXIMATE SOLUTION FOR THE PROPER SELF-ENERGY

In analogy to the calculations on the single chain \cite{16} we restrict ourselves to small external forces applied on the ends of the crosslinked chain. Therefore we make the same ansatz for the proper self-energy as in the case of the single chain \cite{16}. The only difference is that the proper self-energy in this paper is a function depending on the replica-space wave vector $\hat{q}$:

$$M(\hat{q}) = a_0 + a_1 \hat{q}^2 + O(\hat{q}^4)$$  \hspace{1cm} (23)

To start with let $M_r(\hat{q}) = M(\hat{q}) - M(0)$. Then $M_r(\hat{q})$ is given by

$$M_r(\hat{k}) = 4\pi b z^2 (m + 1) \left[ \int_{q(0)} \frac{1}{(q(0) + k(0))^2 + \kappa^2} \frac{1}{\mu + \frac{\kappa^2}{6} (q(0), k(1..m))^2 + M_r(q(0), k(1..m))} \right]$$

$$- \int_{q(0)} \frac{1}{(q(0))} \frac{1}{(q(0))^2 + \kappa^2} \frac{1}{\mu + \frac{\kappa^2}{6} (q(0))^2 + M_r(q(0), 0)} \right]$$  \hspace{1cm} (24)

where $\mu_r = \mu + a_0$. Since we assume a replica symmetric solution for the proper self-energy, we consider the second derivative of $M_r(\hat{k})$ with respect to $k(0)$ at vanishing replica-space wave vectors $\hat{k}$, which yields a result for $a_1$. The details of calculation are exactly the same as for the single chain \cite{16}.

$$a_1 = \frac{2bz^2}{3l^2 \pi \mu_0} \left[ 1 + O\left( \frac{\kappa l}{\sqrt{\mu_0}} \right) \right]$$  \hspace{1cm} (25)

This result is valid for $\beta \hat{q}/\kappa < 1$ or in terms of the force $\beta f/\kappa < 1$. For details see \cite{16}. It is important to mention that the right hand side of equation (26) diverges even in the limit $m \to 0$. If we take the limit $m \to 0$, then the proper self-energy $a_0$ is given by

$$a_0 = \frac{2bz^2 (m + 1)}{\pi} \int_0^\infty dq \frac{q^2}{q^2 + \kappa^2} \frac{1}{\mu + a_0 + \frac{\kappa^2}{6} q^2 + a_1 q^2}$$

$$+ \frac{2z_c (\mu + a_0)^{m+\frac{1}{2}}}{2^3 (m+1) \pi \Gamma \left( \frac{3}{2} (m+1) \right) \left( \frac{\kappa^2}{6} + a_1 \right)^{m+\frac{1}{2}}} \int_0^\kappa \frac{d\hat{q}}{\sqrt{\hat{q}}} \frac{q^2 (1 + 3m \ln(q))}{1 + \hat{q}^2}$$  \hspace{1cm} (27)

Note, that the integration-variable $\hat{q}$ is dimensionless. The integrals in Eq. (27) can be performed and the result is:

$$a_0 = \frac{12bz^2 (m + 1) \sqrt{6(\mu + a_0)}}{\pi (6\mu + 6a_0 - 6\kappa^2a_1 - \kappa^2 l^2) \sqrt{\mu l^2 + 6\mu a_1 + a_0 l^2 + 6a_0 a_1}} \arctan \left( \frac{\kappa l^2 + 6a_1}{\sqrt{6(\mu + a_0)(l^2 + 6a_1)}} \right)$$

$$- \frac{3bz^2 (m + 1) \kappa}{6\mu + 6a_0 - 6\kappa^2a_1 - \kappa^2 l^2}$$  \hspace{1cm} (28)
charge

\[ z \]



\[ \text{Here we reintroduced the force \( \beta \) acting on the ends of the chain.} \]

\[ \begin{align*}
&+ \frac{2z_c(\mu + a_0)^{\frac{m+1}{2}}}{\Gamma(m+1)} \left[ \frac{1}{\kappa} \sqrt{\frac{l^2/6 + a_1}{\mu + a_0}} \left( 1 - 3m + 3m \ln \left( \frac{\kappa}{\sqrt{\frac{l^2/6 + a_1}{\mu + a_0}}} \right) \right) 
- \arctan \left( \frac{\kappa}{\sqrt{\frac{l^2/6 + a_1}{\mu + a_0}}} \right) - \frac{3m\pi}{4} \ln \left( 1 + \frac{\kappa^2(l^2/6 + a_1)}{\mu + a_0} \right) 
+ \frac{3mi}{2} \left( \text{dilog} \left( i\kappa \sqrt{\frac{l^2/6 + a_1}{\mu + a_0}} \right) - \text{dilog} \left( i\kappa \sqrt{\frac{l^2/6 + a_1}{\mu + a_0}} \right) \right) \right] 
\end{align*} \]

where \( \text{dilog}(x) \) is the dilogarithmic function, defined as:

\[ \text{dilog}(x) = \int_1^x \frac{\ln(t)}{1-t} \, dt \] (29)

Starting from expression (28) we neglect terms of order \( \mathcal{O}(m^2) \). As we consider weakly charged networks we only take into account terms up to the order of \( z^2 \). Terms of higher order in the charge \( z \) are neglected. We are interested in the long ranged limit of the Debye-Hückel potential. So the next step is to make a series expansion with respect to small \( \kappa \) including terms of order \( \kappa^3 \). Note, that the sequence of the series expansions with respect to the monomer charge \( z \) and the inverse Debye-Hückel screening length \( \kappa \) is important. For physical reasons it does not make sense to consider first the long ranged limit before calculating the weakly charged case. The result of this analysis is:

\[ a_0 = \frac{bz^2(m + 1)\kappa}{\mu + a_0} \left( \frac{2}{\pi} - \frac{1}{2} \frac{\kappa^2 l^2}{12(\mu + a_0)} + \frac{2\kappa^2 l^2}{9\pi(\mu + a_0)} \right) \]

\[ + \frac{\kappa^3 z_c}{\pi^2(\mu + a_0)} \left( \frac{m\gamma}{4} + m \ln(\kappa/\sqrt{\pi}) - \frac{2m}{3} + \frac{1}{6} + \mathcal{O}(m^2, z^4, \kappa^4) \right) \] (30)

with \( \gamma \approx 0.5772157 \) Euler’s constant. Neglecting terms of order \( a_0^2 \) in Eq. (31) leads to a linear equation in \( a_0 \) which can be solved. The result for \( a_0 \) is:

\[ a_0 = \frac{\kappa}{\mu} \left( \frac{2bz^2(m + 1)}{\pi} - \frac{bz^2(m + 1)}{2} + \frac{2\kappa^2 l^2 b^2 z^2(m + 1)}{9\pi} - \frac{2\kappa^2 l^2 b^2 z^2(m + 1)}{12\mu} \right) \]

\[ - \frac{\kappa}{\mu} \left( \frac{2\kappa^2 z_c m}{3\pi^2} - \frac{\kappa^2 z_c \gamma m}{4\pi^2} - \frac{\kappa^2 z_c m \ln(\kappa/\sqrt{\pi}) m}{2\pi^2} - \frac{\kappa^2 z_c}{6\pi^2} \right) \] (31)

Since the expansion coefficients \( a_0 \) and \( a_1 \) for the proper self energy \( M(\hat{k}) \) due to the calculation above are known approximately it is possible to write down the Greens function formally:

\[ \hat{G}(\hat{k}, \mu, z_c) = \frac{1}{\mu + a_0 + \frac{z_c}{6} \hat{k}^2 + a_1 \hat{k}^2} \] (32)

V. RESULTS

From Eq. (32) the grand canonical partition function in replica space under the influence of an external constant force \( f \) acting on the ends of the chain can be calculated:

\[ \Xi_m(\mu, z_c, f) = \hat{G}(\hat{k}^{(0)}, \hat{k}^{(1)} - i\beta f, \ldots, \hat{k}^{(m)} - i\beta f, \mu, z_c) |_{\hat{k} = 0} \]

\[ = \frac{1}{\mu + a_0 - \frac{z_c^2}{6} \beta^2 f^2 - a_1 \beta^2 f^2} \] (33)

Here we reintroduced the force \( f \) according to Eq. (10). The monomer chemical potential and the fugacity of cross-links which parameterize the grand canonical partition function in Eq. (33), should be expressed in terms of the parameters \( N_{\text{tot}} \) and \( N_c \). According to Panyukov and Rabin [19] the expression \( F_m(N_{\text{tot}}, N_c) \) can be calculated by the method of steepest descent in the thermodynamic limit \( N_{\text{tot}}, N_c \to \infty \):

\[ F_m(N_{\text{tot}}, N_c)/k_B T = -\ln(\Xi(\mu, z_c) - N_{\text{tot}} \mu + N_c \ln z_c) \] (34)
Consequently the fugacity \( z_c \) of cross-links and the chemical potential \( \mu \) of monomers can be obtained by minimizing the right-hand side of Eq. (34):

\[
N_{\text{tot}} = - \frac{\partial \ln \Xi_{m=0}(\mu, z_c)}{\partial \mu} \tag{35}
\]

and

\[
N_c = \frac{\partial \ln \Xi_{m=0}(\mu, z_c)}{\partial \ln Z_c} \tag{36}
\]

in the limit of a vanishing replica index \( m \). It can be shown from Eqs. (35) and (36), that:

\[
\bar{N} = \frac{3z_c\mu^2\kappa^3}{8\kappa^3z_c^2b^2\pi + 36\kappa\mu b^2z_c^2\pi + 3\kappa^3z_c\mu - 18\mu^3\pi^2 - 9\kappa\mu b^2z_c^2\pi^2 - 3\kappa^3b^2l^2\pi^2} \tag{37}
\]

where \( \bar{N} \) is defined as the crosslink-density \( N_c/N_{\text{tot}} \) of the network. Neglecting the charge per monomer \( z \) the crosslink density reads:

\[
\bar{N} \approx \frac{3z_c\mu^2\kappa^3}{3z_c\mu^3 - 18\pi^2\mu^3} \tag{38}
\]

Since \( \bar{N} \) is a positive number \( z_c \) has to be large enough, namely:

\[
z_c > \frac{6\pi^2\mu^2}{\kappa^3} \tag{39}
\]

At this point we make a series expansion of Eq. (38) with respect to small \( \mu \) neglecting terms of order \( \mu^2 \) which is possible, because \( \mu \) is connected with the crosslink-density \( \bar{N} \), that is assumed to be small:

\[
\bar{N} = \mu + O(\mu^2) \tag{40}
\]

To calculate \( z_c \) we make the following ansatz according to inequality (39) with a positive \( y \):

\[
z_c = \frac{6\pi^2\mu^2}{\kappa^3} + y \tag{41}
\]

Now we substitute this ansatz for \( z_c \) and the result for the chemical potential \( \mu \) into Eq. (37) and calculate \( y \). Therefore we again make a series expansion with respect to small monomer charges \( z \) neglecting terms of order \( z^4 \).

\[
\bar{N} = \frac{2\pi^2\bar{N}^2(9\pi^2\bar{N} - 36b\bar{N}\pi)z^2}{\kappa z^2} + \frac{2\pi^2\bar{N}^2}{\kappa z^2} \left( 3\bar{N} + \frac{3b^2l^2\pi^2 - 8bz^2l^2\pi}{y} \right)
\]

\[
+ \frac{3b^2\bar{N}\pi^2 - 12bz^2\bar{N}\pi}{\kappa z^2} + \bar{N} + \frac{3b^2l^2\pi^2 - 8bz^2l^2\pi}{3y} + O(z^4)
\]

This equation can be solved with respect to \( y \). Again neglecting terms of order \( z^4 \) and only taking into account the leading term with respect to a small inverse Debye-Hückel screening length \( \kappa \) the result is:

\[
y = \frac{3\pi(4 - \pi)b^2}{\kappa^2} + O(z^4, \kappa^0) \tag{43}
\]

Now we can write down the result for \( z_c \) which reads:

\[
z_c = \frac{6\pi^2\bar{N}^2}{\kappa^3} + \frac{3\pi(4 - \pi)b^2}{\kappa^2} \tag{44}
\]

Moreover it can be shown [19], that the conformational free energy of the network is given by:

\[
F(N_{\text{tot}}, N_c, f) = -k_BT \frac{\partial \ln \Xi_m(\mu, z_c)}{\partial m} \bigg|_{m=0} \tag{45}
\]
If the free energy $F$ is known the force-size relationship is the simply calculated by the derivative of $F$ with respect to the external force $f$:

$$\langle R \rangle = -\frac{\partial F(N_{tot}, N_c, f)}{\partial f} \quad (46)$$

After having inserted the results for $z_c$ and $\mu$ in the force size relationship, expanded for small charges $z$ neglecting terms of order $z^4$ and higher, since only weakly charged networks are stable, and considered only terms of leading order with respect to small $\kappa$ the force-size relationship reads:

$$\langle R \rangle = \left( \frac{b z^2}{3 N^2 \pi \kappa} + \frac{l^2}{6 N} \right) \beta f + O(z^4, \kappa) \quad (47)$$

This is again the result for the small deformation regime. The result describes a Hookian law for the force extension relation and defines the elastic modulus of the polyelectrolyte network. Note further that this result is valid for small forces, i.e. $\beta f/\kappa < 1$. Therefore the modulus for the small screening and the low deformation regime of the network reads:

$$G = \left( \frac{b z^2}{3 N^2 \pi \kappa} + \frac{l^2}{6 N} \right)^{-1} \quad (48)$$

which is the central result of our paper. The modulus depends on the density of the crosslink and on the Debye screening parameter. Thus both contributions enter in a significant way. Most striking is that part of the modulus stemming from the interactions, which depends on the crosslink density $N$ squared.

**VI. DISCUSSION**

In the previous sections we analyzed the force size relationship of a polyelectrolyte network, which was made of a very long crosslinked chain. The method presented, which was developed for a single chain \[17\], is in replica-space applicable to polyelectrolyte networks. We considered the network in the long ranged limit of the Debye-Hückel potential.

An important result is that for small forces and weakly charged networks the response to an external force $f$ is proportional to $f$. The modulus $G$ depends on $N^2$, where $N$ is the crosslink-density, which is a surprising result in contrast to classical considerations on networks, where the modulus is proportional to $N$ \[23\]. To discuss the result for the modulus in more detail let us discuss it in the form

$$G^{-1} = \beta \left( \frac{l^2}{6 N} + \frac{b z^2}{3 N^2 \pi \kappa} \right) \quad (49)$$

The modulus consists of two terms. The first part, $G_N \propto k_B T \bar{N}$, of the modulus is the term corresponding to classical rubber elasticity. It is proportional to the temperature and to the crosslink density. This corresponds to the usual entropy elasticity of connected flexible chains. The factor $1/6$ appears only from the choice of the special representation of the network, i.e., one macroscopic chain and has no specific physical meaning. The second part, i.e., $G_1 \propto (k_B T N^2 \kappa)/(b z^2)$ stems purely from the interactions. It is not entropy elastic, since the Bjerrum length $b$ and the Debye screening parameter depends on temperature. The overall temperature dependence is given by $G_1 \propto T^{3/2}$. Since both parts have a really distinguished temperature dependence, they can be separated experimentally in a clear way. Moreover the strong difference in the crosslink dependence, $G_N \propto \bar{N}$, and $G_1 \propto \bar{N}^2$ allows also a clear experimental separation.

It is interesting to note that the two different terms combine in as two springs in series, one entropic one (the rubber network) and another energetic one, coming from the interactions. If the strength of the springs is very different, naturally in such systems always the weaker dominates the main elasticity.

It is important to realize that interactions and elasticity interplay in a clear way. Our results yield then the conclusion that the Flory assumption, i.e., adding the different parts of elastic and interaction part, is no longer valid in these systems. These approximations are perhaps on a level of the random phase approximation, but clearly the field theoretical variational technique used in the present paper is beyond perturbative methods used so far in neutral \[14\] networks and polyelectrolyte \[12\] networks. The Flory - Rehner assumption normally uses for the total free energy of the network $F = F_{\text{elastic}} + F_{\text{int}}$, i.e., the addition of the elastic and the interaction part of the free energy \[14\]. In this hypothesis the \textit{bare} expression of the elastic free energy is used which is proportional to the crosslink density $\bar{N}$. Here
we have shown that due to crossterms the elastic modulus becomes renormalized by the interactions. Thus we claim that the simple addition theorem is no longer valid.

The next step will be to apply the results from the present paper to several experimental situations. In a subsequent and less detailed paper we will study the effects on swelling and the thermodynamic behavior. In following works we will also study the free energy functional at smaller length scales which will provide informations on the scattering behavior of the network. This is especially important when deformation processes on different scales are considered. Extensions to entangled systems will also become important within this context. And finally, we have to revisit the Debye - Hückel approximation. So far, we had assumed that the counterion are freely distributed, an assumption which corresponds to the Debye - Hückel approximation. Of course, correlation effects will change the picture and these higher order effects must be the subject of subsequent studies.

ACKNOWLEDGMENTS

The authors wish to thank Firma Stockhausen GmbH, D-47705 Krefeld, Germany for financial support.

[1] C. Tanford, Physical Chemistry of Macromolecules, (Wiley, New York) 1961
[2] K. S. Schmitz in Macro-ion Characterization K. S. Schmitz, Ed. (American Chemical Society, Washington) 1994
[3] T. Liverpool and M. Slapper, Europhys. Lett., in press, (1997)
[4] S.F. Edwards, J. Phys. A 8, 1670, (1975)
[5] M. Doi and S.F. Edwards, The Theory of Polymer Dynamics, Oxford, Clarendon Press, 1986
[6] J. P. Flory, Principles of Polymer Chemistry, Cornell University Press, Ithaca, 1959
[7] J. P. Flory and J. Rehner Jr., J. Chem. Phys., 11, 512, (1943)
[8] R. W. Brotzmann and B. E. Eichinger, Macromolecules, 16, 1131, (1983)
[9] R. C. Ball and S. F. Edwards, Macromolecules, 13, 748, (1980)
[10] T. A. Vilgis and J. Wilder, submitted to Computational and Theoretical Polymer Science, 1997
[11] A. R. Khokhlov, S. G. Starodubtzev, V. V. Vasilievskaya; Advances in Polymer Science 109, 123, (1993)
[12] M. Muthukumar, in A. Baumgärtner and C. Picot (eds), Molecular Basis of Polymer Networks, Springer Verlag, Heidelberg, 1989
[13] M. Rubinstein, R. H. Colby, A. V. Dobrynin, J. F. Joanny, Macromolecules 29, 398, (1996)
[14] S.F. Edwards and T.A. Vilgis, Rep. Prog. Phys., 51, 243 (1988)
[15] P.-G. de Gennes, P. Pincus, R. M. Velasco, F. J. Brochard; Phys. Fr. 37, 1461, (1976)
[16] P. Haronska, J. Wilder and T. A. Vilgis, J. Phys. II, 7, 1273, (1997)
[17] R. T. Dean and S. F. Edwards, Philos. Trans. R. Soc. London Ser. A, 280, 317, (1976)
[18] M. Stevens and K. Kremer, J. Chem. Phys 103 (4), 1669, (1995)
[19] S. Panyukov and Y. Rabin, Statistical Physics of Polymer Gels, Physics Reports 269, 1, (1996)
[20] S. F. Edwards, Proc. Phys. Soc. London, 85, 613, (1965)
[21] S. K. Ma Modern theory of critical phenomena, (Addison -Wesley) 1995
[22] J. F. Joanny and L. Leibler, J. Phys. 51, 545, (1990)
[23] L. R. G. Treloar, The Physics of Rubber Elasticity (Oxford Clarendon Press, Oxford) 1975.
[24] T. A. Vilgis and R. Borsali, Phys. Rev. A 43, 6857 (1991)