Research on Friction and Wear Properties of Rubber Composites by Adding Glass Fiber during Mixing

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Abstract: GF/rubber composites have sound insulation characteristics, heat resistance, good corrosion resistance, and high mechanical strength. The compounding machine’s long working hours will inevitably wear the metal on the end face of the compounding machine. The wear of the end face metal will increase the gap between the chamber and the end face, which will lead to material leakage, reduce the mixing effect, and eventually affect the performance of GF/rubber composites. To ensure the implementation of GF/rubber composites, it is necessary to study the frictional wear behavior of GF/rubber composites on metals. In this paper, the effect of blending rubber with different amounts of GF on the frictional wear of metal on the end face was analyzed from the perspective of the formulation process, and the ratio of corrosion wear and abrasive wear was calculated for the first time.

Keywords: GF/rubber composite; end metal; wear amount; wear ratio

1. Introduction

Glass fiber is a kind of material with excellent performance, which has many types, low price, and high mechanical strength. It is widely used in the reinforcement of polymer materials. The composites obtained by adding glass fibers have excellent properties. It has been reported that the composite material of glass fiber and rubber has good comprehensive properties and can improve its anti-skid performance [1–5]. As it is widely used in many fields, glass fiber is increasingly valued. Glass fibers are commonly used as reinforcement materials in composite materials, electrical insulation materials, thermal insulation materials, circuit substrates, etc. An overview diagram of GF is shown in Figure 1.

Li Hong [6] reviewed the development of continuous glass fibers in recent years, focusing on high-performance glass fibers. In particular, the article presents recent advances in the characterization of glass structures using NMR spectroscopy, Raman spectroscopy, and molecular dynamics simulation techniques.

Yang Chuanqi [7] used the mechanical blending method to prepare glass fiber (GF)/natural rubber (NR)/butadiene rubber (BR) composites and studied the influence of glass fiber partially replacing silica on the vulcanization properties, mechanical properties, wear resistance, and skid resistance of composites. The performance of the filler was compared with that of silica and nanometer attapulgite. The results show that adding glass fiber can shorten the curing time and increase the compound’s hardness, tear strength, and wet friction coefficient. When the amount of glass fiber is 6 phr, the comprehensive performance of the binder is higher than that of silica and nanometer attapulgite filler.

Wang Wenjie [8] used glass fiber (GF) as a filler and nitrile butadiene rubber (NBR) as a matrix to prepare NBR/GF composites by the mechanical blending method and studied the effect of the amount of glass fiber on the properties of the composites. It was found that with
the increase of the glass fiber dosage, the vulcanization rate of the compound decreased gradually, and the tensile strength and 100% fixed elongation stress of the vulcanized compound increased first and then decreased. The maximum tensile strength and 100% fixed elongation stress were increased by 33% and 550%, respectively.

Figure 1. SEM image of GF. (a) magnification is 10 μm, (b) magnification is 5 μm).

Chen Jiabin [9] prepared glass fiber/silicon rubber composite materials, studied the failure characteristics of glass fiber/silicon rubber composite materials by the tensile test, analyzed the stress/strain curve of the composite material, and used the finite element method to numerically simulate the composite material. The study found that the glass fiber/silicone rubber composite specimen mainly manifested as a fiber debonding fracture with fiber pulling out under the action of tensile load.

Lv Fuling [10] prepared nano-magnesium hydroxide/glass fiber/silicon rubber composites by mechanical blending. The dispersion state of nano-fillers in the silicone rubber matrix was investigated, and the changing laws of physical and mechanical properties, heat resistance, and flame-retardant properties of composites with different filler dosages were compared. The results show that additional magnesium hydroxide and glass fiber dosage ratios can improve silicone rubber’s physical and mechanical properties. Among them, tensile strength, tear strength, compression set, and hardness have been enhanced. Combining the two fillers improves the composite’s thermal stability and flame retardant properties to a certain extent.

Hao Qiangqiang [11] prepared glass fiber/nitrile rubber composites by the mechanical blending method, studied the effect of glass fiber content and length on the wear resistance and mechanical properties of the composites, and observed the wear surface of the composites with a scanning electron microscope. The results show that the optimum addition mass of chopped glass fibers in the nitrile rubber matrix is 3 phr, and the length is 6 mm. Under these conditions, the prepared composites have the best wear resistance and comprehensive mechanical properties, and the wear surface is the flattest and is smooth.

The preparation of GF/rubber composites is carried out in the compounding machine. With the development of GF/rubber composites, the use time of the compounding machine, which is the equipment for preparing GF/rubber composites, has increased dramatically, bringing about the wear and tear of the compounding machine. This paper studies the frictional behavior of GF/rubber composites with different ratios and end metals during the mixing process.
2. Experiment

2.1. Instruments

The experimental equipment used in this experiment is shown in Table 1.

Table 1. Experimental instruments.

| Experimental Instruments | Country of Production | Manufacturers                        |
|--------------------------|-----------------------|---------------------------------------|
| Hake mixer               | China                 | Qingdao University of Science and Technology |
| BL-6157 double-roll rolling mill | China             | Dongguan Baolun Precision Testing Instrument Co. |
| DisperGRADER Dispersion Meter | America          | Alpha Corporation                     |
| LEXT OL5500 3D laser measurement microscope | Japan            | Products of Olympus Corporation       |
| CSM-friction and abrasion tester | Switzerland     | Tribometer Corporation                |
| RPA2000 Rubber Processability Analyzer | America          | Alpha Corporation                     |
| ZT-2588S Steam Generator | China                 | Zhiteng Company                       |

2.2. Formulation

Table 2 shows the formulae. To calculate the silylation reaction index, adding the C1 group without TESPT as a control is necessary.

Table 2. Formulation.

| Raw Material | C1 | C2 | C3 | C4 | C5 | C6 |
|--------------|----|----|----|----|----|----|
| TSR20        | 15 | 15 | 15 | 15 | 15 | 15 |
| BR9000       | 25.5 | 25.5 | 25.5 | 25.5 | 25.5 | 25.5 |
| RC2557S      | 82 | 82 | 82 | 82 | 82 | 82 |
| Silica115MP   | 45 | 45 | 45 | 45 | 45 | 45 |
| GF            | 2 | 2 | 2 | 2 | 2 | 2 |
| ZnO           | 0 | 0 | 1 | 3 | 5 | 7 |
| TESPT         | 0 | 0 | 6 | 6 | 6 | 6 |
| DPG           | 0.8 | 0.8 | 0.8 | 0.8 | 0.8 | 0.8 |
| SAD           | 2 | 2 | 2 | 2 | 2 | 2 |
| 4020          | 2 | 2 | 2 | 2 | 2 | 2 |
| CZ            | 1.8 | 1.8 | 1.8 | 1.8 | 1.8 | 1.8 |
| S             | 1.3 | 1.3 | 1.3 | 1.3 | 1.3 | 1.3 |

The following were used: Butadiene Rubber 9000 (BR9000), China Hainan Natural Rubber Industry Group Co., Ltd., (Haikou, China); Polymeric Styrene Butadiene Rubber (RC2557S), Dongguan Futai Rubber Trading Co., Ltd., (Dongguan, China); Natural Rubber (TSR20), China Hainan Natural Rubber Industry Group Co., Ltd., (Haikou, China); Bis-[γ-(triethoxysilyl)propyl]tetrasulfide (TESPT), Antioxidant 4020, Zinc Oxide ZnO, Stearic Acid SAD, 1,3-Diphenylguanidine (DPG), accelerator CZ, and sulfur S, products of China Henan Longji Chemical Co., Ltd., (Puyang, Chian); Glass Fiber (GF), Wanlong Composite Materials Co., Ltd., (Qingdao, Chian); Silica115MP has a specific surface area of 115 m²/g, 7–40 nm particle size, and solid powder, Solvay Silica Co., Ltd., (Brussels, Belgium); GF diameter is 12–23 µm, length is 0.2–0.6 mm, and the surface is not treated.

2.3. Mixing Procedure

The GF/rubber composites were obtained after 5 min by mixing according to the mechanical compounding process in Table 3. The obtained GF/rubber composite was over-rolled in the BL-6157 double-roller mill, and the setting was set to a width of the double-roller of 8 mm, and the GF/rubber composite specimens with a thickness of 8 mm were obtained. The obtained GF/rubber composite specimens were cut according to the die of CSM to obtain friction specimens with a diameter of 100 mm and a thickness of 8 mm.
Table 3. Mechanical blending process.

| Time (s) | T (°C) | Ingredients                        |
|---------|--------|------------------------------------|
| 0       | 70     | BR9000, RC2557S, TSR20             |
| 40      |        | TESPT, GF, DPG, SAD, 4020, ZnO, 1/2 Silica |
| 70      |        | 1/2 Silica                         |
| 150     | 120    | Sweep                              |
| 240     | 135    | Sweep, Sampling                    |
| 300     | 145    | Discharge                          |

2.4. Test Methods

2.4.1. Rubber Processing Properties

This test was performed on the RPA 2000, and the test conditions were set as follows. The silica Durocher Stabilized Cool Down scan test conditions were 0.01 Hz scan frequency, 0.28–40% scan strain range, and 60 °C temperature, and the dynamic modulus $G'$ curve with strain was obtained [12–17].

(1) Payne effect: The phenomenon in which the elastic modulus of a filler compound decreases as the strain amplitude decreases [18–20].

(2) Silanization reaction index: Characterize the extent to which the silanization reaction proceeds. The calculation method is shown in Table 4. The test principle of the degree of silanization reaction is shown in Figure 2.

\[
X = \frac{\text{Area of silylation zoon}}{\text{Area of the largest silylation region}} = \frac{\int G'_{REF}(05) - \int G'_{S}(05)}{\int G'_{REF}(05) - \int G'_{S}(06)}
\]

Table 4. Testing method.

| Stage | Frequency/Hz | T (°C) | Strain  | Test Items |
|-------|--------------|--------|---------|------------|
| 1     | 0.1          | 60     | 0.28%   | $G'$ (01)  |
| 2     | 1            | 60     | 0.28–40%| $G'$ (02)  |
| 3     | 1            | 60     | 0.28–40%| $G'$ (03)  |
| 4     | 0.1          | 60     | 0.28%   | $G'$ (04)  |
| 5     | 1            | 60     | 0.28–40%| $G'$ (05)  |
| 6     | 1            | 60     | 0.28–40%| $G'$ (06)  |

Figure 2. Distribution of silanization reaction degree.
2.4.2. Friction Test

The Hake mixer is shown in Figure 3. The CSM friction and wear testing machine was used to conduct friction experiments.

![Hake internal mixer](image1)

**Figure 3.** Hake internal mixer.

The CSM was parameterized before the experiment. The rotation speed of the CSM was the same as that of the Hake mixer, which was set to 70 r/min, the pressure was set to 5 N, and the friction time of the CSM was set to 30 min. Previous studies have shown that the most severe wear occurs at the end of the experiment, so the CSM temperature was set to 150 °C. The CSM friction and wear tester is shown in Figure 4 [21–25].

![CSM friction and wear tester](image2)

**Figure 4.** CSM friction and wear tester.

(3) Metal surface observation: The metal grinding head was placed on the stage and the LEXT OLS5000 3D microscope was used to emit a laser to scan the metal surface to obtain metal surface data [26–28].

(4) Dispersion: The rubber sample was tested with a DisperGRADER disperser, and the dispersion value was obtained through equipment analysis.

3. Mechanistic Study

3.1. Silylation Reaction Mechanism

The silanization reaction is divided into two steps, as shown in Figures 5 and 6.
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Figure 4. CSM friction and wear tester.

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3. Mechanistic Study

3.1. Silylation Reaction Mechanism

The silanization reaction is divided into two steps, as shown in Figures 5 and 6. Figures 5 and 6 show the silanization process. Due to limited technical conditions, the quality of ethanol and water produced during blending cannot be measured. To ensure the reliability of the test results, friction and wear experiments were carried out on CSM. According to the proportion of the silanization reaction, the high-temperature ethanol–water mixed vapor was sprayed onto the metal surface. The mixing conditions of the mixer were qualitatively simulated [29–37].

3.2. Friction Mechanism between GF/Rubber Composite and Metal

The forms of friction between GF/rubber composites and metals are mainly classified as abrasive and corrosive wear, and this study focuses on these two aspects:

(1) SiO₂ particles have the characteristics of extremely easy agglomeration, while SiO₂ particles are complex and irregular surfaces, so SiO₂ particle aggregates are the leading cause of abrasive metal wear [38–44].

(2) The high-temperature ethanol–water mixed vapor will cause corrosion wear to the metal.

4. Experimental Results

4.1. Dispersion Analysis

4.1.1. Payne Effect

The stress–strain curves of GF/rubber composites with different ratios are shown in Figure 7, and the Payne effects of GF/rubber composites with different ratios are shown in Table 5.
With the increase of GF content in the rubber matrix, the Payne effect of the composites when the added GF exceeded 3 phr.

Table 5. Payne effect.

| Formula | C1     | C2     | C3     | C4     | C5     | C6     |
|---------|--------|--------|--------|--------|--------|--------|
| Payne effect | 939.19 | 875.45 | 700.99 | 547.01 | 1099.81| 1339.08|

The data from Figure 7 were extracted in Table 5. From Table 5, it can be seen that the Payne effect of GF/rubber composites without TESPT addition was more significant. With the increase of GF content in the rubber matrix, the Payne effect of the composites first decreased and then increased. When the GF content was 3 phr, the Payne effect of the composite was the lowest. The Payne effect of the GF/rubber composite increased sharply when the added GF exceeded 3 phr.

GF has a unique spatial structure, and its surface can adsorb SiO$_2$ particles in the rubber matrix. This promotes the dispersion of SiO$_2$ particles in the rubber matrix and reduces the number of SiO$_2$ particle agglomerates. Therefore, the Payne effect of GF/rubber composites decreases with GF addition. GF is a spatial filamentous structure, which is easy to agglomerate with each other in the rubber matrix, which hinders the dispersion of SiO$_2$ particles. Meanwhile, the total surface area decreases after GF accumulation, and the adsorption of SiO$_2$ particles is weakened. All of this leads to an ever-increasing Payne effect.

4.1.2. Dispersion Degree of GF/Rubber Composite

From Figure 8 and Table 6, it can be seen that the GF/rubber composite had the lowest dispersion and the best dispersion when the content of GF was 3 phr. Adding too much or too little GF will increase the distribution of GF/rubber composites. This corresponds to the Payne effect of GF/rubber composites.

Figure 7. Stress–strain curves of GF/rubber composites with different ratios.

Table 5. Payne effect.

| Formula | C1     | C2     | C3     | C4     | C5     | C6     |
|---------|--------|--------|--------|--------|--------|--------|
| Payne effect | 939.19 | 875.45 | 700.99 | 547.01 | 1099.81| 1339.08|

Figure 8. Cont.
The silylation reaction indices of GF/rubber composites with different ratios are shown in Figure 9. The silanization index is in Table 7.

It can be seen from Figure 9 that when the amount of GF added is 3 phr, the silanization index of the composite material is close to 1, and the silanization reaction is most advanced at this time. The silanization reaction of the composites with the remaining amount of GF was decreased. This is consistent with the previous analysis of the Payne effect.

Figure 8. Dispersion images (the scale bar is 100:1).

Table 6. Dispersion values.

| Rubber Compounds | C1   | C2   | C3   | C4   | C5   | C6   |
|------------------|------|------|------|------|------|------|
| Dispersion       | 5.28 | 5.97 | 6.58 | 7.78 | 6.61 | 4.69 |

4.2. Silylation Reaction Index

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Figure 9. Cont.
4.3. Silylation Reaction Index

Table 7. Silylation reaction index.

| Compounding Rubber | C2   | C3   | C4   | C5   | C6   |
|--------------------|------|------|------|------|------|
| Silylation reaction index | 0.339 | 0.532 | 0.819 | 0.331 | 0.081 |

4.3. Different Ratios of GF/Rubber Composite with Metal Friction and Wear

4.3.1. Friction Coefficient

It can be seen from Figure 10 that the friction coefficient when the composite material rubs against the metal first decreased and then increased. The friction coefficient of GF/rubber composites was the smallest when the GF addition was 3 phr. It can be seen from Figure 10 that the friction coefficients of the experimental groups added with 0, 1, 5, and 7 phr GF increased by 22.4%, 18.5%, 20.8%, and 29.1%, respectively, relative to the experimental group with 3 phr of GF added. With the increase of GF content in the rubber matrix, SiO$_2$ particles are continuously adsorbed on the surface of GF, which promotes the dispersion of SiO$_2$ particles and reduces the number of SiO$_2$ particle agglomerates in the rubber matrix. SiO$_2$ particles have high physical hardness. The SiO$_2$ particle aggregates have complex and irregular shapes, making the composite material have serious friction against the metal during the friction process, and the friction coefficient is significant. When the content of GF in the rubber matrix exceeded 3 phr, the aggregates of SiO$_2$ particles in the rubber matrix continued to increase, which made the friction between the composite material and the metal intensify during the friction process and the friction coefficient increased.
4.3.2. Metal Surface Topography

Figure 11 shows the height profile of the metal grinding head before and after sanding.

![Height profile of metal grinding head before and after friction.](image)

Figure 11. Height profile of metal grinding head before and after friction.

It can be seen from Figure 11 that a large number of contour peaks were ground after the metal surface of the C2 group was rubbed. There were many pits on the metal surface, and the wear was more serious. A part of the height profile peaks of the metal surface before and after friction in the C3 group was smoothed, which was reduced compared to the C2 group. After the metal surface of the C4 group was rubbed, a few height peaks were filed, the number of pits appeared less, and the wear was lighter. The metal surface of group C4 had more height peaks being ground after friction, the metal surface profile changed significantly, and the wear was more serious. The height profile of the metal surface before and after friction in the C6 group changed significantly, and a large number of height peaks were filed.

4.3.3. Metal Wear

From Figure 12, it can be seen that with the increase of GF addition, the wear amount of GF/rubber composites on metal decreased and then increased, and the GF/rubber composite with a 3 phr GF addition had minor wear on metal. There was no adsorption of GF on SiO₂ particles in the composite material without GF added, and the dispersion of SiO₂ particles could not be promoted, so the metal wear was severe.
Figure 12. Metal wear volume. ((A) is the volume of the metal before and after friction, and (B) is the volume change).

During the mixing process of the composites with 1 part of GF added, some SiO$_2$ particles were adsorbed on the surface of GF, which reduced the number of SiO$_2$ particle agglomerates in the composites. The SiO$_2$ particle aggregates have complex and irregular shapes, making the composites produce severe friction with the metal during the friction process. With the reduction of SiO$_2$ particle agglomerates in the composites, the abrasive wear of the composites to metals was gradually weakened. With the increase of the degree of the silanization reaction, the amount of high-temperature ethanol–water vapor increased, which increased the proportion of corrosion wear.

When 3 phr of GF was added, the surface area of GF that could adsorb SiO$_2$ particles increased, and GF could adsorb more SiO$_2$ particles, which better promoted the dispersion of SiO$_2$ particles and further reduced the number of silica aggregates. Therefore, the GF/rubber composite with 1 phr of GF and the GF/rubber composite with 3 phr of GF had minor wear on the metal. As can be seen from Figure 12, the experimental group added 3 phr of GF, relatively. The wear amount of the experimental groups with 0, 1, 5, and 7 phr of GF added increased by 60%, 38.7%, 116%, and 146.7%, respectively.

4.3.4. Wear Ratio

The abrasive wear amount of the metal grinding head is shown in Figure 13.

Figure 13. Metal wear volume without spraying high-temperature ethanol–water vapor mixture.

No high-temperature steam was sprayed during this experiment, and there was no corrosion and wear of metal, so the data measured in this experiment were the amount
of abrasive wear. As can be seen from Figure 13, the experimental group added 3 phr of GF, relatively. The abrasive wear of the experimental groups with 0, 1, 5, and 7 phr of GF increased by 66.7%, 42%, 124.6%, and 165.2%, respectively. Compared with the above experiment, the wear ratio was obtained, and the wear ratio image is shown in Figure 14.

![Figure 14. The proportion of abrasive wear and corrosion wear.](image)

The increase in the degree of silanization reaction increased the amount of high-temperature ethanol–water mixed vapor, which increased the proportion of corrosive wear. When the GF addition amount was 3 phr, the contact area between GF and SiO$_2$ particles reached the maximum, and the SiO$_2$ particles were best dispersed. Therefore, the GF/rubber composite with a 3 phr GF addition had the most significant degree of silanization reaction and the highest percentage of corrosion and wear. Compared with the experimental groups with 0, 1, 5, and 7 phr of GF added, the corrosion wear rate with 3 phr of GF increased by 115.2%, 56.4%, 116.8%, and 748%, respectively.

4.3.5. Roughness

From Figure 15, it can be seen that the amount of roughness change on the metal surface decreased and then increased with the increase of GF addition, which is related to the adsorption of GF to SiO$_2$ particles. As can be seen from Figure 15, the experimental group added 3 phr of GF, relatively. The roughness variation of the experimental groups with 0, 1, 5, and 7 phr of GF increased by 191.8%, 97.4%, 103%, and 232%, respectively. With the increase of GF addition, the number of silica aggregates decreased. Silica is more complex and wears seriously on metal, so the surface roughness of metal gradually decreased. When the content of GF in the composite matrix exceeded 3 phr, the agglomeration of SiO$_2$ particles in the composite material increased continuously, and the wear of the composite material to the metal increased. This resulted in severe wear on the metal surface and increasing roughness variation.
The proportion of abrasive wear and corrosive wear.

Figure 14. The proportion of abrasive wear and corrosive wear.

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Figure 15. Metal surface roughness.

5. Conclusions

This study found that GF has a large surface area, and its surface can adsorb nano-SiO2 particles. With the increase of GF content in the composite matrix, the degree of the silanization reaction of the composite material first increased and then decreased. The amount of high-temperature ethanol–water mixed vapor corresponds to the degree of the silanization reaction. The proportion of abrasive wear of GF/rubber composites to metals decreased and then increased, and the proportion of corrosive wear increased and then decreased. With the increase of GF addition, the contact area between GF and SiO2 particles gradually increased, the dispersion of SiO2 particles was enhanced, the number of silica aggregates decreased, and the amount of GF/rubber composites to metal wear decreased. When the addition amount of GF was 3 phr, the GF/rubber composite had the lowest wear on metal. When the addition amount of GF exceeded 3 phr, there was a large amount of accumulation between GF, the contact area between GF and SiO2 particles gradually decreased, the dispersion of SiO2 particles decreased, the number of silica aggregates increased, and the wear amount of GF/rubber composites on metal increased. In the long run, the strengthening and transforming infrastructure in the Middle East and Asia-Pacific will increase the demand for GF. This study plays a vital role in reducing the wear of the internal mixer, improving the life of the internal mixer, and ensuring the performance of GF-modified rubber.

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