Chemical and Molecular Variations in Commercial Epoxide Photoresists for X-ray Lithography

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Featured Application: The characterization of negative photoresists before the application on LIGA process can be a key factor to save time, money and enhance the reproducibility of the main optical components fabricated to be used in techniques such as Talbot–Lau Interferometry.

Abstract: The quality of high aspect ratio microstructures fabricated by deep X-ray lithography is highly dependent on the photoresist material used and the process parameters applied. Even with photoresists more suitable to this process, it is common to face defects in the final optical components, such as in case of X-ray gratings. The gratings need to be fabricated with critical dimensions on a sub-micrometer and micrometer scale, with periods of few micrometers and heights of hundreds of micrometers to be used in X-ray imaging techniques such as Talbot–Lau Interferometry. During the fabrication process, these features lead to challenges such as mechanical stability, homogeneity, and defect-free grating patterns. Usually, an epoxy-based negative photoresist is used in X-ray lithography, which needs to account for the shrinkage that takes place during polymer crosslinking in order to avoid defects in the final pattern. Nowadays, photoresist material still lacks certain suitable properties (chemical and mechanical) to fabricate gratings of high quality and with acceptable reproducibility. This work presents the results of TGA, FTIR, and MALDI-TOF analysis made on photoresists commercially available and suitable for X-ray lithography. The photoresists presented different profiles regarding the solvent content and oligomers composition, and in the case of some samples, there were high amounts of non-epoxidized oligomers.

Keywords: X-ray lithography; negative tone photoresist; microfabrication; Talbot-Lau Interferometry

1. Introduction

1.1. Deep X-ray Lithography (DXRL)

Deep X-ray lithography (DXRL), which is the first step of the LIGA (a German acronym for “Lithographie, Galvanoformung, Abformung”) process, is used to manufacture high aspect ratio (HAR) micrometer and sub-micrometer structures out of polymer [1]. Depending on the structural requirements and other boundary conditions (e.g., tolerable exposure time), either a negative or a positive tone photoresist is used. Polymethylmethacrylate (PMMA) is the most suitable positive tone photoresist, delivering the highest structural quality, but it suffers from low sensitivity, resulting in
prolonged exposure time especially for several hundred micrometer high structures [2,3]. Chemically amplified negative tone photoresists are much more sensitive but show some structural deficiencies due to shrinkage during polymerization after exposure. It is evident that the quality of the products fabricated by the LIGA process is highly dependent on the raw materials used for patterning, as well as on their chemical behavior during processing. Hence, it is necessary to have strict control of the process parameters [4–12]. The photoresist layer to be patterned is fabricated by spin coating a photoresist to the desired thickness in a range from 5.0 µm to several millimeters, depending on the application. The photoresist mixture contains at least a polymer/oligomer with photosensitivity, a catalyst, a solvent, and occasionally some additives [13,14]. Negative photoresist solutions are usually based on epoxy resins, because this resin has mainly been used for microstructure fabrication (MEMS), optical sensors, micro-pieces of watches, radiation sensors, optical fiber components, conductive microstructures, photonic crystals, etc. [15–17]. Compared to other materials, this resin is relatively cheap; the microstructures exhibit excellent properties, such as high mechanical resistance, thermal and chemical stability, and excellent transparency [14,17–19].

Despite many advances in microstructure fabrication, unknown factors affecting the quality and reproducibility of the HAR microstructures still exist; some of these defects are shown in Figure 1 (all images were made after the development step).

Figure 1a shows, in the region protected from the X-rays (the black area), a crosslinked material abnormality (“cobweb”). This cobweb effect can occur due to the high sensitivity of the photoresist to secondary irradiation (e.g., photo electrons). The lamellae deformation shown in Figure 1b is a direct result of the shrinkage/swelling properties of the photoresist during polymerization. Depending on the mechanical properties of the crosslinked photoresist, this may even result in cracks (Figure 1d). A thermal mismatch of the crosslinked photoresist with the substrate and the lower mechanical stability or capillary forces in the drying process may result in structural collapse (Figure 1c). Bubbles caused during the baking and/or exposure and/or post-exposure bake steps conglomerate in prominent clusters. The residual photoresist will then create a thin layer on top of the structures, which remains solid after the development process, affecting the mechanical stability of the HAR microstructures and
blocking the electroforming process (Figure 1e). Figure 1f indicates problems in the adhesion between
the photoresist and the substrate.

1.2. Epoxy-Based Photoresists

The negative tone photoresists are based on a resin known as SU-8, where the molecular structure
is photosensitive. It was developed initially by IBM, which patented it in 1989; afterward, it was
continued to be produced by Shell Chemical (EPON SU-8 epoxy resin) [20,21]. The molecular structure
of the resin present in the photoresists (Figure 2) has three main organic groups in the chemical
structure: the epoxy groups (responsible for crosslinking), the aryl rings, and the methylene bridges
that bind the polymer chain. The polymer has an excellent ability to interact with its own chains;
however, when mixed with other polymeric materials, low adhesion and interaction are observed.
It displays good adhesion to metallic surfaces, surfaces pretreated with plasma, and with hydroxyl
groups (OH-) bonded to the surface. The photoresist exhibits high optic transparency for wavelengths
above 360 nm and high sensitivity to ultraviolet (UV) wavelength (350–400 nm); it is inert to several
solvents, bases, and acids and presents thermal-stability after curing (crosslinking reaction) [14,22].

![Figure 2. Polymer structure for most of the negative tones of photoresists.](image)

The photoresists also contain at least more two components, a solvent and a catalyst to initiate
the crosslinking process. The solvent is normally used to tune the viscosity of the polymer; the most
applied solvents are cycle ketones or lactones (pentanone or gamma butyrolactone). The catalysts,
also known as photo acid generators (PAG), are divided into two groups: ionic and non-ionic.
The families of triarylsulfonium and diaryliodonium hexafluoride salts are largely applied to initiate
the crosslinking reaction (curing process) of epoxy resins, such as the tris(4-methoxyphenyl)sulfonium
hexafluoroarsenate [23]. These salts form strong acids to start the crosslinking process, releasing a
proton to initiate the reaction in the epoxy groups [23]. The crosslinking reaction can be initiated in
acidic or basic conditions; the type of catalyst defines the mechanism. In most cases, the epoxides are
polymerized in acidic conditions with the help of high-energy radiation [24], taking as an example the
salts mentioned above. The catalysts are present in a range of 1–10% in weight at the composition of
the photoresist [25,26].

Although the mechanical properties of photoresists and their processing parameters are very
well described in the literature [2,3,6,7,9,11,12,27–32], less attention has been given to the chemical
structure and the effects caused by the catalysts, additives, and solvents in the photoresists’ mixtures.
The investigation of the chemical properties concerning the quality of the final structures fabricated by
DXRL can help to improve the quality of the structures. It also helps to establish a method to evaluate
the quality of the photoresist before its use and to estimate its structural quality. This work aims to
characterize the negative photoresists used in the standard fabrication of microstructures by DXRL.
and to evaluate their chemical compositions to optimize the quality of the final structures taking into account the process parameters.

In this work, two types of negative tone photoresists were evaluated. The first type was named EPR-S1, of which two lots were evaluated (“a” and “b”). The second negative photoresist used to achieve HAR structures was named EPR-S2, of which eight different lots were evaluated (“a” to “h”).

2. Materials and Methods

To characterize the samples of photoresists (specifically, epoxy-based negative tones of photoresists, customized to X-ray lithography and commercially available), the following techniques were used: Fourier-transform infrared (FTIR) spectroscopy, thermogravimetric analysis (TGA), and matrix assisted laser desorption ionization time of flight (MALDI-TOF) mass spectroscopy. All the reagents and chemicals were acquired through the Sigma-Aldrich (Sigma-Aldrich, Darmstadt, Germany).

2.1. Thermogravimetric Analysis

The tests were done in an oxidative atmosphere (synthetic air) in a NETZSC TG-209 (Erich NETZSCH GmbH & Co. Holding KG, Selb, Germany) using about 30–50 mg of the material. The temperature range of the operation was 25–800 °C with a heating ramp of 10 °C·min⁻¹.

2.2. Fourier-Transform Infra-Red Spectroscopy

The analyses by FTIR spectroscopy were performed using a spectrometer Bomem of Hartmann & Braun, model B-100 (ABB, Zürich, Switzerland). The samples were characterized in the liquid phase (approximately 5 mg of the photo resin solution), where a film was performed under a potassium bromide crystal. The spectral information was obtained in a range between 500 and 4000 cm⁻¹ with a resolution of 4 cm⁻¹ and 32 scans.

2.3. Matrix Assisted Laser Desorption Ionization Time of Flight Mass Spectroscopy

The spectra were recorded on a KRATOS AXIMA Performance MALDI instrument (Shimadzu, Nakagyo-ku, Japan). A nitrogen laser was applied as irradiation source with a wavelength of 337 nm. The analyses were carried out with the following characteristics: positive polarity, flight path-linear, mass-high –20 kV voltage, 100–150 pulses per spectrum. The samples were prepared as follows: a range of 4–6 mg of photoresist was diluted in 1 mL of acetone. The matrix was prepared separately with 10 mg of 2,3-Dihydroxybenzoic acid (DHB) and 4 mg of NaCl; both also solubilized in 1 mL of acetone. Thus, the sample and matrix were mixed, and 1 µL of this mixture was placed on the MALDI target.

3. Results and Discussion

3.1. FTIR Spectra Evaluation

Figure 3 shows an example of the FTIR spectra of the photoresist EPR-S1_b, while the Figure 4 present the main organic groups characterized. All of the other samples were analyzed separately and afterward plotted together (see Figure 5).
Figure 3. Fourier-transform infrared (FTIR) spectra of sample EPR-S1_b.

There are eight specific important regions in the FTIR spectra (Figure 3). The region in 3475 cm\(^{-1}\) is a vibration typically assigned to hydroxyl groups (OH\(^-\)), and the region at 2930 cm\(^{-1}\) confirms the presence of methyl, methylene, and aromatic hydrogens in the structure of the sample. The regions in 1798 and 1735 cm\(^{-1}\) are assigned for carbonyl of ketones, acids, esters, and lactones groups. The regions of 1600 and 829 cm\(^{-1}\) are assigned to alkenes, and the region in 1247 cm\(^{-1}\) is assigned to the organic group ether. Figure 4 resumes the organic fractions present in the sample.

Figure 4. Organic groups identified in the FTIR spectra analysis.

The same profile in the FTIR spectra was observed in the other samples (Figure 5). There were only significant differences between the EPR-S1 and EPR-S2 photoresists, where the band for carbonyl (range of 1700–1800 cm\(^{-1}\)) was less intense and a second band was observed in the case of the EPR-S1 series. Also, this second band was delocalized for the left side (smaller wavelength).

The organic groups that characterize the polymer are present in all the samples of EPR-S1 or EPR-S2. However, other organic groups were also identified, indicating the presence of other compounds (e.g., solvents, catalysts, etc.). The assignment in 3475 cm\(^{-1}\) (hydroxyl groups) can indicate the non-total reaction in the epoxidation step of the synthesis (see Section 3.3. MALDI-TOF Analysis).
3.2. TGA Analysis

The solvent content of the photoresists was evaluated by TGA, performing the oxidative process of the material until 800 °C. Figure 6 shows the results for the resin EPR-S1 and is used as the example to explain the main changes during the degradation process.

The TGA graph (Figure 6) can be divided into three main parts: the region of solvent evaporation (room temperature until 220 °C), the region of the crosslinking process (220–320 °C), and the region of degradation (320–650 °C) [33]. The derivative curve of mass loss allows us to evaluate the first event of a mass loss in the graphs, which is due to solvent evaporation (blue region in Figure 7).

Figure 7 shows the TGA curves for all of the characterized photoresists. There are differences in the solvent content of up to 5.5% approximately. Usually, the solvent is applied to obtain the
same viscosity parameter. As the viscosity is a physical-chemical parameter directly related to the molecular composition [34], this difference can be initially interpreted as evidence of differences in the composition of the photo resin solutions. Thus, a difference in the solvent content in the same type of photoresist means that the molecular structures are slightly different.

![Figure 7. TGA of photo resins.](image)

For the series EPR-S2, the measurements showed solvent content between 22.66% and 28.15%. The two samples of EPR-S1 series also demonstrated considerable differences in the solvent content. These differences are not critical depending on the final application. However, specifically in the microlithography process, this can be a significant difference between the raw materials that were obtained by the same supplier and theoretically should have the same composition.

### 3.3. MALDI-TOF Analysis

Although the MALDI is a technique that can be interpreted only in a semi-quantitative approach and the mechanism of ionization is not yet known entirely, the ionization techniques, such as electrospray ionization (ESI) and MALDI, have the advantage of analyzing intact polymer/oligomer molecules, avoiding fragmentation reactions. It is a soft ionization technique and the polymer/oligomer is not influenced by the solution preparation methods. Once the technique is carried out in a solid state (the solvent must be evaporated in the sample preparation), the sample preparation has a short time. It has considerably high sensitivity, and also, MALDI is a suitable technique to characterize factors such as the polydispersion, data of the repeat units, impurities, and additives in synthetic polymers. It can also be applied to a considerable range of polar and non-polar polymers with an extensive range of molecular masses [35–39].

In previous times, before the ionization sources coupled with mass spectroscopy, there were not many techniques able to evaluate mass distribution in polymer samples. The secondary ion mass spectrometry (SIMS) and field desorption (FD) could give information about the mass distribution but with a considerable limitation, in that the polymers must be soluble and with maximum molecular mass near to $10 \times 10^3$ Da·mol$^{-1}$ [40]. NMR spectroscopy and chromatography techniques can provide information about the molecular structures and molecular size, but the distribution concerning chain sizes cannot be fully understood with these methods for a large range of synthetic polymers [41].

To observe the composition of the resins with respect to molecular size, the photoresists were evaluated by MALDI-TOF in view of their oligomer content. There is no influence of the solvent
content and catalyst, the range of molar size of these compounds is low compared to the range applied in this analysis.

The major composition of oligomers present in the samples corresponds to the dimer (region of 715 Da) and trimer (region of 1067 Da) molecules, as Figure 8 shows.

![Figure 8. Matrix assisted laser desorption ionization time of flight (MALDI-TOF) of the photo resin samples.](image)

Although the major compounds are the same, there are distinct profiles in the samples (e.g., in the mass profile of sample EPR-S2_f are several mass peaks that are not observed in the spectra of the other samples). These differences are due to the synthesis procedure applied. Observing with more details, the range of low-mass spectra (Figure 9, region between 500 and 800 Da), the samples can be separated into groups: the samples EPR-S2_c and EPR-S2_f; samples EPR-S2_b and EPR-S2_e; samples EPR-S2_a, EPR-S2_g, and EPR-S2_h; the sample EPR-S2_d; and finally, the EPR-S1 series. The samples EPR-S2_c and EPR-S2_f presented several mass peaks in the low-mass region, meaning that the polymer chains vary (high polydispersity) or/and had several secondary products. The samples EPR-S2_b and EPR-S2_e are composed mostly of dimers and trimers of the same size approximately. Also, among the samples EPR-S2_a, EPR-S2_g, and EPR-S2_h, the spectral profile is similar. The compositions of sample EPR-S2_d and the EPR-S1 series consist mostly of dimer molecules. These differences in molecular size distribution might result directly in different physical-chemical properties.

Moreover, in the spectra, it is also possible to observe several peaks that represent the oligomers in different compositions regarding the bonds with the epoxy groups. The theoretical possibilities are monomer two possible peaks; dimer four possible peaks (see Figure 10 using the dimer as an example); trimer six possible peaks; tetramer eight possible peaks; pentamer ten possible peaks; and hexamer twelve possible peaks.
These possibilities do not consider the secondary products that can be generated during the epoxidation step. As a result, the molar mass can vary around 30%. Table 1 shows the mass possibilities as a function of the oligomer’s size in the photo resin composition:

| Oligomer | ED0  | ED1  | ED2  | ED3  | ED4  | ED5  | ED6  | ED7  | ED8  | ED9  | ED10 | ED11 | ED12 |
|----------|------|------|------|------|------|------|------|------|------|------|------|------|------|
| Monomer  | 228.29 | 284.36 | 340.32 | -    | -    | -    | -    | -    | -    | -    | -    | -    | -    |
| Dimer    | 468.59 | 524.66 | 580.72 | 636.79 | 692.85 | -    | -    | -    | -    | -    | -    | -    | -    |
| Trimer   | 708.89 | 764.96 | 821.02 | 877.09 | 933.15 | 989.22 | 1045.28 | -    | -    | -    | -    | -    | -    |
| Tetramer | 949.20 | 1005.26 | 1061.32 | 1117.39 | 1173.45 | 1229.52 | 1285.58 | 1341.65 | 1397.71 | -    | -    | -    | -    |
| Pentamer | 1213.49 | 1343.62 | 1269.55 | 1399.68 | 1325.62 | 1455.75 | 1381.68 | 1511.81 | 1437.74 | 1567.87 | 1493.81 | -    | -    |
| Hexamer  | 1429.80 | 1485.86 | 1541.93 | 1597.99 | 1654.06 | 1710.12 | 1766.19 | 1822.25 | 1878.31 | 1937.38 | 1988.43 | 2046.51 | 2102.57 |

**Figure 9.** Region of 450–1200 Da in the MALDI-TOF of the photo resin solution samples.

**Figure 10.** Mass variation in function of the epoxidation degree.

Table 1. Mass peaks as a function of the oligomer size versus epoxidation degree.

ED = Epoxidation degree (see Figure 10 as example); Mass peaks unit = Daltons.
To evaluate the spectra, it is necessary to take into account the influence of the matrix dissociation during the analysis method. The MALDI-TOF has the advantage of avoiding the fragmentation of analyte molecules when a light-absorbing matrix is used. Thus, only intact molecules are observed in the mass spectrometer. For polymer samples, it is possible to identify all the absolute sizes in the distribution after the polymerization step. However, it is common that the matrix has an interaction with the sample, leading to the formation of adducts [35,40–44]. causing some differences in the Dalton values expected. As mentioned in the methods section, in this work DHB and sodium chloride were used as matrices. Therefore, it could be possible that the compounds have increased Dalton values. As an example, the main possible adducts for the monomer are presented in Figure 11.

The adduct formation can occur with all molecules in the photoresist. In the spectra, it will be possible to observe a minimum of three different mass peaks for the same structure (the pure oligomer, the adducts with sodium, and the adducts with DHB). Although adducts with DHB do not occur typically, the mass of these adducts was also calculated. Not all the peaks will always be observed because the interaction with the matrix is a random process. In Table 2, the mass peaks that could be observed in the different monomers are listed. Dimers, trimers, tetramers, pentamers, and hexamers were identified in the samples mainly with the sodium adduct.

![Figure 11](image-url)  
**Figure 11.** (a) Dihydroxybenzoic acid (DHB) adduct with the monomer, (b) Sodium adduct with the monomer.

Correlating the data of Tables 1 and 2, it is possible to identify the main oligomer fractions in the photoresists. The most important mass peaks are 491, 533, 715, 1067, 1364, and 1419 Da, representing sodium adducts coming from the matrix solution, as Figure 12 shows.
Table 2. Mass peaks as a function of the oligomer size versus adducts with sodium and DHB.

| Epoxidation Degree | ED0   | ED1   | ED2   | ED3   | ED4   | ED5   | ED6   |
|--------------------|-------|-------|-------|-------|-------|-------|-------|
| Monomer            | aNa   | aDHB  | aNa   | aDHB  | aNa   | aDHB  | aNa   |
| Monomer            | 251.28| 382.41| 307.35| 438.48| 363.31| 494.44|       |
| Dimer              | 491.58| 622.71| 547.65| 678.78| 603.71| 734.84| 715.84|
| Dimer              |       | 694.72| 550.71| 781.84| 706.84| 837.91| 818.91|
| Trimer             | 731.88| 863.01| 787.95| 919.08| 945.11| 1071.21| 1052.21|
| Trimer             |       | 897.01| 753.01| 985.12| 900.12| 1126.21| 1107.21|
| Tetramer           | 972.19| 1103.32| 1028.25| 1159.38| 1084.31| 1215.44| 1196.44|
| Pentamer           | 1212.49| 1343.62| 1268.55| 1399.68| 1324.62| 1455.75| 1436.74|
| Pentamer           |       | 1408.64| 1332.64| 1463.75| 1388.75| 1519.84| 1499.84|
| Hexamer            | 1452.79| 1583.92| 1508.85| 1639.98| 1564.92| 1696.05| 1677.05|
| Hexamer            |       | 1583.92| 1508.85| 1639.98| 1564.92| 1696.05| 1677.05|

| Epoxidation Degree | ED7   | ED8   | ED9   | ED10  | ED11  | ED12  |
|--------------------|-------|-------|-------|-------|-------|-------|
| Oligomer           | aNa   | aDHB  | aNa   | aDHB  | aNa   | aDHB  |
| Oligomer           | 1364.64| 1495.77| 1420.7| 1551.83|       |       |
| Tetramer           | 1604.94| 1736.07| 1661  | 1792.13| 1848.2| 1773.13|
| Pentamer           | 1845.24| 1976.37| 1901.3| 2032.43| 1960.37| 2091.5|
| Hexamer            |       |       |       |       |       |       |

Mass peaks unit = Daltons; aNa = Mass peaks values of sodium adducts; aDHB = Mass peaks values of dihidroxybenzoic acid adducts.
Observing the spectra (Figures 8 and 9), there are significant concentrations of the dimer ED4 (715/716 Da) at all the samples. Some samples showed in their composition, the dimer ED0 (491/492 Da). In this case, it can be considered as a contaminant in the mixture because this structure does not have epoxy groups to support crosslinking. In all samples, considerable concentrations of the trimer ED6 (1068/1069 Da) were present in their composition. Also, all the samples showed traces of the tetramers ED7 (1365/1366 Da) and ED8 (1420/1421 Da). An important characteristic was that the samples EPR-S2_c and EPR-S2_f contained several traces of secondary products (oligomers not epoxidized entirely) in the region of 500–1000 Da, which affects the physical-chemical properties. Thus, although the samples theoretically have the same composition, due to the findings, it is probable that their physical-chemical properties are not the same. Also, a significant characteristic to observe is the influence of the incomplete oligomers in the photoresists. The spectra in Figures 13 and 14 present the amount of the oligomers and the incomplete oligomers for the different samples.
The samples presented different compositions mainly in the dimers and trimers (Figure 13) and in the non-epoxidized dimers (dimers ED0, Figure 14) (e.g., EPR-S2_b and EPR-S2_c are the most concentrated with respect to dimers ED0; the sample EPR-S2_c shows almost four times higher concentration than the other samples). The samples of EPR-S1 contain a small amount of dimer ED0. It is important to mention that the concentration of the dimers ED0 in the samples is comparable to the epoxidized oligomers with respect to the intensity values.

4. Conclusions

Negative photoresists based on epoxy resins and applied in deep X-ray lithography were evaluated by FTIR and presented similar profiles concerning the organic groups in the mixture.
However, between the series EPR-S1 and EPR-S2, a small difference in the region of the carbonyl groups was observed. By TGA, different amounts of solvent in the same lot of photoresist were observed. Although this result does not seem to be significantly important for standard applications, this difference can be significant in X-ray microlithography fabrication, where structures with sizes in the micron range and accuracy in the sub-micron range has to be achieved. By MALDI-TOF analyses, it was possible to observe different profiles regarding the oligomer content. Some photoresists (EPR-S2_b and EPR-S2_c) presented significant amounts of non-epoxidized groups. This evaluation can be a key to achieve better quality structures in microfabrication. In the future, it is important to correlate the differences in solvent and oligomer content with the physical-chemical properties of the photoresists and the characteristics of the microstructures.

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