Density Functional Study of the L1$_0$-αIrV Transition in IrV and RhV

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Abstract

Both IrV and RhV crystallize in the αIrV structure, with a transition to the higher symmetry L1$_0$ structure at high temperature, or with the addition of excess Ir or Rh. Here we present evidence that this transition is driven by the lowering of the electronic density of states at the Fermi level of the αIrV structure. The transition has long been thought to be second order, with a simple doubling of the L1$_0$ unit cell due to an unstable phonon at the R point (0$\frac{1}{2}\frac{1}{2}$). We use first-principles calculations to show that all phonons at the R point are, in fact, stable, but do find a region of reciprocal space where the L1$_0$ structure has unstable (imaginary frequency) phonons. We use the frozen phonon method to examine two of these modes, relaxing the structures associated with the unstable phonon modes to obtain new structures which are lower in energy than L1$_0$ but still above αIrV. We examine the phonon spectra of these structures as well, looking for instabilities, and find further instabilities, and more relaxed structures, all of which have energies above the αIrV phase. In addition, we find that all of the relaxed structures, stable
and unstable, have a density comparable to the \( L1_0 \) phase (and less than the \( \alpha \text{IrV} \) phase), so that any transition from one of these structures to the ground state will have a volume change as well as an energy discontinuity. We conclude that the transition from \( L1_0 \) to \( \alpha \text{IrV} \) is probably weakly first order.

**Keywords:** Structural Phase Transitions, Jahn-Teller, Electronic Structure, Density Functional Theory, Ordered Intermetallic Alloys

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### 1. Introduction

A major goal of computational condensed matter physics is the determination of structural properties for compounds. Once the structure has been determined, other properties, e.g., strength, ductility, electronic properties (including superconductivity), magnetism, etc., can be determined. Several methods exist to attack this problem. One possibility is to exhaustively search an experimental database of known structures\[1\], determining the low energy structures for each composition of the target materials. Other methods use first-principles calculations on a small set of target structures to determine parameters which can be used to predict properties of more complex systems. These include the Cluster Expansion Method\[2, 3, 4, 5, 6\], tight-binding parametrization methods\[7, 8\], and atomistic potential methods\[9\]. Methods of the first type can be combined with methods of the second type for more thorough searches\[10\].

These programs, can, however, only be implemented with a thorough knowledge of the behavior of compounds. Exhaustive searches of a database require an extensive database to search, including all structures which are known, or thought likely, to form for a target system. Parameterized methods may not be able to reach all regions of phase space, and also need a database of structures for testing\[11, 12\]. For both methods, then, it is useful to look at less-common structures found in nature.

The \( \alpha \text{IrV} \) structure, shown in Figure\[1\], is found in nature only in the prototype compound\[13, 14\] and its neighbor in the periodic table, \( \text{RhV} \)[15, 16]. This orthorhombic structure, space group Cmmm (\#65), can be viewed as a doubled unit cell distortion of the \( \text{CsCl} \) structure\[17\] or the tetragonal \( L1_0 \) structure, space group P4/mmm (\#123)\[18\]. Indeed, both the \( \text{IrV} \) and \( \text{RhV} \) phase diagrams\[19\] show the \( L1_0 \) structure as the ground state for Vanadium-poor \( (x = 40 - 48\%) \) \([\text{Ir,Rh}]_{1-x} \text{V}_x\), and Vanadium-rich \( \alpha \text{IrV} \) is known to
Figure 1: The αIrV structure. The box indicates the boundaries of the full orthorhombic unit cell. In the L1₀ or (CsCl) structure the Iridium atoms would be in a tetragonal (cubic) arrangement around each of the Vanadium atoms.

transform to L1₀ at temperatures above 506°C.[18]

The seemingly straightforward transition pathway from L1₀ to αIrV was analyzed by Chen and Franzen[18] in the context of Landau theory: First, double the tetragonal L1₀ unit cell along the y and z directions, corresponding to a phonon at the R (0112) point in reciprocal space. Second, construct the primitive base-centered orthogonal unit cell, and allow the two Ir atoms in the cell to move opposite each other along the z-axis, while the V atoms move along the y-axis. The resulting structure is symmetrically-equivalent to that of αIrV and, assuming a second-order phase transition, relaxes into the αIrV structure.

This simple picture is not quite accurate. A true second order transition, as described by Chen and Franzen, requires a continuous energy path from L1₀ to αIrV. Thus the energy of the L1₀ structure would be lowered
by making an infinitesimal displacement of the type described above, corresponding to a phonon instability at the R point. We have performed first-principles density functional theory calculations for the phonon frequencies at the R point, both by the frozen phonon method\cite{20} and by linear response\cite{21, 22, 23}. We find that all of the modes here have real frequencies, i.e., they do not lead to an instability.

Thus the transition from the L1_0 phase to the αIrV phase does not proceed through the simple unit-cell doubling picture described above. Instead, as we shall show, the L1_0 phase has vibrational instabilities in another region of the Brillouin zone. This paper will discuss the structures arising from these instabilities, and explore the possibility that a second-order transition might go through one of those phases.

In Section 2 we show the results of density functional theory calculations for the energy and electronic structure of the αIrV and L1_0 phases of IrV, RhV, and some neighboring compounds. We show that in IrV and RhV the L1_0 structure has a relatively high electronic density of states at the Fermi level, and so the transition to αIrV has Jahn-Teller character\cite{24}.

In Section 3 we look at the phonon spectra of the αIrV and L1_0 phase. Not surprisingly, we find that the αIrV phase has no phonon instabilities. We do, however, find that the L1_0 structure is vibrationally unstable on and near the line (x = 1/2) in reciprocal space.

In Section 4 we use the frozen-phonon method on the unstable phonon modes to search for new structures which have lower energy than L1_0. Such a search is computationally bound, so we looked at all unit cells with eight atoms or less, and two structures with 32 atom unit cells. In this range we find no instability which relaxes to the αIrV phase, however it is not impossible that searching through larger unit cells would find such an instability, in which case the transition would indeed be second order. We do find several new structures, some of which have apparently never been seen in nature, and one which was previously known\cite{16}.

In Section 5 we discuss our results, and thoughts on the order of the L1_0-αIrV transition.

2. Energetics and Electronic Structure of the L1_0 and αIrV Phases

All computations were made using the Kohn-Sham\cite{25} formulation of Density Functional Theory\cite{26} with the Perdew-Burke-Ernzerhof\cite{27} generalized gradient approximation. Depending on our needs, we used the Vienna
Figure 2: Energy/volume curves for IrV and RhV in the L1$_0$ and αIrV structures, determined from VASP and QE calculations as discussed in the text. For ease of comparison we have set the minimum energy of the L1$_0$ phase to zero.

*ab initio* Simulation Package (VASP)\textsuperscript{28, 29, 30} with projector augmented-wave (PAW) potentials\textsuperscript{31}, or the *Quantum Espresso* (QE) package\textsuperscript{32} with the supplied ultra-soft pseudopotentials for Ir, Rh, and V.

We used rather large plane wave cutoffs of 350 eV in VASP and 540 eV in QE to ensure convergence. Similarly, we used 12 × 12 × 8 Γ-centered k-point meshes in both the L1$_0$ and αIrV structures, leading to 140 and 215 points, respectively, in the irreducible Brillouin zones. We summed over electronic states using a Fermi-Dirac distribution\textsuperscript{33} with a temperature of 65 meV (0.005 Rydbergs). These values give well-converged results.

We show our results for IrV and RhV in Table 1 and Fig. 2. In both cases we see that the VASP PAW potentials and the QE ultrasoft pseudopotentials are in excellent agreement with one another, and in good agreement with experiment, within the usual errors of Density Functional theory in the generalized gradient approximation. In both compounds the αIrV state is below the L1$_0$ state by approximately 55 meV/formula unit, in agreement with experiment.

From experiment\textsuperscript{18} we know that L1$_0$ is the preferred high temperature structure for αIrV. Using the COMSUBS routine from the ISOTROPY\textsuperscript{34, 35} package, we find that an orthorhombic distortion of the L1$_0$ structure reduces the symmetry from space group P4/mmm to Cmm, the space group of the αIrV structure. It is plausible to argue\textsuperscript{18} that this lowering of symmetry is the pathway for the L1$_0$ → αIrV phase transition. The symmetry-breaking character of the transition is evident from Fig. 3 where we plot the electronic density of states of both phases near the Fermi level.
Table 1: Equilibrium lattice constants (in Å), atomic positions, and equilibrium bulk modulus ($K_0$, in GPa) for the L1$_0$ and αIrV structures of IrV and RhV determined from experiment [13, 15], VASP PAW calculations, and QE ultrasoft pseudopotential calculations. Note that we give the primitive tetragonal lattice parameters for the L1$_0$ structure, rather than the more-common face-centered tetragonal setting, so that when $c/a = 1$ the L1$_0$ structure reduces to the cubic CsCl structure. The final row of the table shows the energy difference between the two phases (meV/formula unit).

| Structure       | Exp.  | VASP | QE   | Exp.  | VASP | QE   |
|-----------------|-------|------|------|-------|------|------|
| L1$_0$ Structure|       |      |      |       |      |      |
| a               | 2.749 | 2.755| 2.762| 2.754 | 2.739| 2.745|
| c               | 3.651 | 3.677| 3.668| 3.599 | 3.660| 3.648|
| $K_0$           | 272   | 277  | 230  |       |      |      |
| αIrV Structure  |       |      |      |       |      |      |
| a               | 5.791 | 5.838| 5.816| 5.78  | 5.849| 5.813|
| b               | 6.756 | 6.759| 6.767| 6.65  | 6.707| 6.725|
| c               | 2.796 | 2.814| 2.823| 2.78  | 2.792| 2.802|
| $K_0$           | 271   | 276  | 227  |       |      |      |
| Ir/Rh (4j)      | 0.22  | 0.216| 0.216| 0.214 | 0.215|      |
| V (4g)          | 0.28  | 0.296| 0.297| 0.296 | 0.297|      |
| $\Delta E$     | 53.6  | 67.0 | 50.4 | 59.7  |      |      |
As we are only interested in the overall behavior of the density of states, we compute these curves by smearing out each eigenvalue found by VASP using a Fermi distribution at a temperature of 5 mRy. We see that at the Fermi level the density of states is twice as large in the L1\textsubscript{0} phase as it is in the \(\alpha\)IrV phase. This is consistent with a Jahn-Teller-like symmetry breaking and phase transition\cite{24}. As an aside, we also note that there is a minimum in the L1\textsubscript{0} density of states just above the Fermi level. This is consistent with the phase diagram of IrV\cite{19}, which shows that Iridium-rich IrV has the L1\textsubscript{0} structure. Assuming the additional Ir replaces V in the L1\textsubscript{0} structure, and using a rigid-band model, we can see how adding Ir to the system would raise the Fermi level and lower the density of states, leading to a more stable L1\textsubscript{0} phase. We find a similar, though less pronounced, state of affairs in RhV, as is also seen in Fig. 3.

We investigated the possibility that the \(\alpha\)IrV structure might be found in compounds neighboring IrV and RhV in the periodic table. The results are shown in Table 2. Except for CoPt, all of the compounds shown here have 14 electrons in the valence band, just as in IrV and RhV. In all cases except CoPt we found that the \(\alpha\)IrV structure was locally stable, that is, if we started from the lattice parameters of the \(\alpha\)IrV, replacing Ir and V by the indicated atoms, and allowed the system to relax, it remained in the \(\alpha\)IrV structure and did not relax to the higher symmetry L1\textsubscript{0} structure. This suggests that the formation of the \(\alpha\)IrV structure is a result of a Fermi level effect associated with 14 electrons in a tetragonal unit cell. We did not check the elastic and vibrational stability of the resulting structure. However the
Table 2: Energy difference, in meV per formula unit between the L1\textsubscript{0} and αIrV structures for the given compounds, as determined by VASP/PAW/PBE calculations. A positive number indicates that the L1\textsubscript{0} structure is lower in energy. An energy difference of zero indicates that the αIrV structure relaxes into the L1\textsubscript{0} structure. Ground state structures are from the ASM Online Handbook\cite{19}.

| Compound | Ground State | L1\textsubscript{0} | D8\textsubscript{b} | L1\textsubscript{0} | αIrV | L1\textsubscript{0} | αIrV |
|----------|--------------|---------------------|---------------------|---------------------|-------|---------------------|-------|
| Δ E      | 0            | 25.3                | 59.0                | 79.9                | -53.6 | 50.1                | 70.6  |

Figure 4: Electronic density of states of the ground-state L1\textsubscript{0} and hypothetical αIrV phases of IrNb, found by smearing the eigenvalues computed by VASP.

L1\textsubscript{0} structure was lower in energy for all compounds except IrV and RhV. Also, as shown in Fig. 4, for these compounds the density of states of the L1\textsubscript{0} structure is very near to that of the αIrV structure, precluding a Jahn-Teller lowering of the energy with decreased symmetry. We therefore conclude that it is unlikely to find the αIrV structure anywhere except in IrV and RhV.

3. Stability of the L1\textsubscript{0} and αIrV Phases

A system is stable, or metastable, only if it has real phonon frequencies for all k-points and modes except the acoustic phonons at the Γ point, which are guaranteed to vanish because of translational symmetry. Since the elastic constants are related to the long-wavelength behavior of the acoustic phonons\cite{37} this implicitly includes the Born stability criteria\cite{38}. This is for-
Figure 5: The base-centered orthorhombic $\alpha$IrV structure. Left: Phonon frequencies IrV at equilibrium, determined using the linear response method with ultrasoft pseudopotentials from the *Quantum Espresso* package. Right: First Brillouin zone. Symmetry lines are described by the notation of Miller and Love.\(^{*}\)

fortunate, since the orthorhombic $\alpha$IrV structure would require us to compute nine elastic constants.

To check the stability of the $\alpha$IrV phase and the possible meta-stability of the L1\(_0\) phase we computed phonon frequencies throughout the respective Brillouin zones using linear response\(^{[21, 22, 23]}\), as implemented in the *Quantum Espresso* package PHon code\(^{[32]}\). Phonon frequencies were computed on a reciprocal space grid ("q-points") using an $8 \times 8 \times 4$ mesh for the L1\(_0\) structure (45 points in the irreducible Brillouin zone) and a $6 \times 6 \times 4$ mesh for the $\alpha$IrV structure (39 points). Linear response calculations at these points yield a reciprocal-space dynamical matrix. These matrices are transformed into a real-space dynamical matrix which can be used to compute phonon frequencies over the entire Brillouin zone.

This approach can lead to aliasing at points off of the q-point mesh. To check that this does not occur we also performed frozen-phonon calculations\(^{[20]}\), wherein we construct a supercell commensurate with the given q-point and measure the change in energy as a function of atomic displacement within the supercell. To do this we used the program FROZSL, part of the ISOTROPY package\(^{[34, 35]}\), with atomic displacements of 0.1 Bohr. Electronic structure calculations were then performed with the same energy cutoff as the original unit cell, and when possible the same k-point mesh, albeit folded back into the smaller Brillouin zone associated with the supercell.

In the following, in interest of saving space, we only show the results for
IrV. Our calculations for RhV show a similar pattern.

Fig. 6 shows the phonon frequencies of the equilibrium structure of αIrV. High symmetry points and lines are labeled according to the convention of Miller and Love[36]. All the phonons have real frequencies, confirming that αIrV is at least a metastable structure for IrV, as suspected. There is no evidence of any aliasing in these calculations.

Figure 6 shows the phonon frequencies of the L1₀ phase of IrV, along symmetry lines labeled according to the convention of Miller and Love[36]. We show results of frozen phonon calculations, showing good agreement between the two techniques, even off the linear response q-mesh. As expected, there are regions of reciprocal space with imaginary phonon frequencies. Surprisingly, this region is not near the R point (011) as described by Chen and Franzen[18], but along and near a line from the midpoint of the U line to the midpoint of the S line, as shown in more detail in the second panel of the figure. We will discuss the implications of this instability in the next section.

4. Searching for Low Energy Structures

The previous section showed that the αIrV structure is the ground state (or, at least, a low-energy metastable state) of IrV, while the L1₀ structure is
unstable to phonons along and near the line $(x \frac{1}{2}, \frac{1}{2}, \frac{1}{2})$ in the Brillouin zone, but not at the R point, where the phonon modes are real, invalidating the second-order phase transition scenario of Chen and Franzen[18]. In this section we will look to see if it is possible to find another continuous transition path from L1$_0$ to αIrV.

In the frozen phonon method, we displace atoms from their equilibrium sites in a supercell consistent with the phonon wave vector and in directions which maintain the symmetry of the phonon mode. The frequency of the mode is then directly related to the square root of the curvature of the energy as a function of atom displacement. A mode with an imaginary frequency will then as a matter of course be related to a supercell calculation which has a negative curvature, leading to supercells with energy lower than the original state. These structures can then be relaxed, leading to new, or at least different, structures.

As an example of this, consider the unstable L1$_0$ phonon at the point $(\frac{1}{4}, \frac{1}{4}, \frac{1}{2})$ on the high symmetry “S” line. The linear response calculations find this phonon frequency to be 81 cm$^{-1}$. To generate a supercell for this calculation we used the FROZSL code from the ISOTROPY package[34, 35]. This package tells us that this point has phonons in three irreducible representations, each with two associated modes. For each representation we run three total energy calculations for displacements of the atoms in different directions, with the maximum displacement of 0.1 Bohr. The energy differences between these structures and the ground state are used to determine the dynamical matrix at this point, and the mode frequencies.

For the equilibrium L1$_0$ parameters found in Table 1, the $S_3$ irreducible representation (in the notation of Miller and Love[36]) has one mode with an imaginary frequency of 74 cm$^{-1}$. (The discrepancy between linear response and frozen phonon calculations is due to the use of different k-point meshes and the anharmonicity of the mode. For our purposes we are only interested in showing that the mode is unstable in both cases, and so will not try to refine the calculations to improve the agreement between the two.)

Diagonalizing the dynamical matrix allows us to find a supercell with displaced atoms which has an energy lower than the L1$_0$ phase. This supercell has space group Cmmm (#65), and with the appropriate choice of origin has Ir atoms at the (4e) and (4h) Wyckoff positions, and V atoms at the (2a), (2b), and (4j) positions. This is crystallographically equivalent to the intermetallic Ga$_3$Pt$_5$ structure [39], and we will refer to it as such. Upon relaxation we find a minimum energy structure which has approximately the
Figure 7: Linear-response phonon spectra of two eight-atom unit cells derived from unstable phonon modes of L1\textsubscript{0} IrV. Left: The metastable Rh\textsubscript{5}V\textsubscript{3} structure. Right: The Amm2 structure with Ir atoms at two (4d) Wyckoff positions and V atoms at four (2b) Wyckoff positions. The imaginary frequencies of the unstable modes are shown as negative frequencies and highlighted in red.

same density as, and an energy 20.5 meV/formula unit below, the relaxed L1\textsubscript{0} structure of Table 1. Note that this is still well above the energy of the αIrV structure.

In a similar fashion, if we look at the phonons at 0\frac{1}{4}1\frac{1}{2}, on the “U” line, we find an imaginary U\textsubscript{3} mode with a frequency of 70 cm\textsuperscript{-1}. This mode leads to another supercell with space group Cmmm, but now the Ir atoms are on the (4e) and (4g) Wyckoff sites, while the V atoms occupy a pair of (4j) Wyckoff sites. The relaxed structure again has a density comparable to L1\textsubscript{0}, but its energy is only 11.0 meV/formula unit below the L1\textsubscript{0} structure minimum. In the discussion below will refer to this structure as Cmmm.

Neither the Ga\textsubscript{3}Pt\textsubscript{5} nor the Cmmm structure will relax to the ground state αIrV structure. However, both structures have imaginary frequency long-wavelength optical mode phonons. If we venture away from the Γ point we will have to deal with frozen-phonon calculations on unit cells having hundreds of atoms, so we will only examine the unstable modes at Γ in both structures.

First consider the imaginary frequency Γ\textsubscript{4} mode of the Ga\textsubscript{3}Pt\textsubscript{5} structure. It is associated with a supercell of space group Amm2 (\#38), with Ir atoms at the (4d) and (4e) Wyckoff sites and V atoms on pairs of (2a) and (2b) sites. This structure is crystallographically equivalent to the Rh\textsubscript{5}V\textsubscript{3} structure\textsuperscript{16,40} and will be referred to by that name. The relaxed structure is 41.1 meV/formula unit below the L1\textsubscript{0} structure, with similar density.
This structure is metastable, as seen in the linear-response phonon spectrum plotted on the left-hand side of Fig. 7.

The second structure, derived from the unstable Γ−3 mode of Cmmm, again with space group Amm2 (#38), puts the Ir atoms on two (4d) sites and the V atoms on four distinct (2b) sites. The relaxed structure has an energy 24.1 meV/formula unit below the L10 structure, again with a similar density. This structure has not been found in the intermetallic literature. It is vibrationally unstable, as can be seen from the phonon spectrum on the right-hand side of Fig. 7.

The four structures Ga3Pt5, Rh5V3, Cmmm and Amm2 are the only structures with eight or fewer atoms in the primitive cell which can be generated by relaxing the unstable phonon modes of the L10 structure. If we wish to examine larger unit cells, we can continue along this line of research ad infinitum, or until we find a supercell where the atoms relax to the ground state αIrV structure, a supercell with energy below the ground state structure, or we run out of computational resources. In most cases, including this one, we will reach the latter limit first. We did compute the relaxed energies of two supercells associated with unstable phonon modes in the Amm2 structure, and one with an unstable mode of the Cmmm structure. These structures had thirty-two atoms in the supercell, and energies below the parent structure but above the metastable Rh5V3 structure.

Tables 3, 4 and 5 summarize all of the calculations discussed here, including the structural derivation, space group, lattice constants, atomic positions, energy, and stability.

5. Discussion

We have shown by direct calculation that the transition from the L10 to the αIrV structure of both IrV and RhV is a result of a Jahn-Teller driven distortion of the high-symmetry unit cell. This distortion does not result from a zone-doubling unstable phonon at the R point of the L10 Brillouin zone, and so the simple Landau theory picture does not hold. This does not completely eliminate the possibility that the transition is second order, for we found an entire region of reciprocal space, in and around the line (x1412), where the L10 structure has vibrational instabilities. We examined several of those instabilities, and found that they lead to numerous new structures, most of which have yet to be seen in ordered intermetallic systems and most of which are vibrationally unstable. The only phase which we found that

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Table 3: Structural, energy, and vibrational stability results for the structures of IrV discussed in this paper. All calculations use the QE pseudopotentials discussed in the text. “Source” indicates the origin of the unit cell, either from experiment or an unstable phonon in the indicated structure. “Atoms” is the number of atoms in the primitive cell. Lattice constants are given in the standard crystallographic convention, e.g. the lattice constants of the full orthorhombic cell are given for the Cmmm and Cmcm space groups. All primitive cells have $\alpha = \gamma = 90^\circ$. The rows “Ir” and “V” give the Wyckoff positions of the atoms. “Volume” is the minimum energy volume of the structure per formula unit, in ($\text{Å}^3$). “Energy” is the energy of the structure below the L1$_0$ structure, in meV/formula unit.

| Structure | L1$_0$ | Cmmm | Cmcm |
|-----------|--------|------|------|
| Source    | Exp.   | $U_3(0\frac{1}{4}\frac{1}{4})$ L1$_0$ phonon | $\Lambda_3(00\frac{1}{4})$ Cmmm phonon |
| Space Group | P4/mmm | Cmmm | Cmcm |
| Atoms | 2 | 8 | 32 |
| a (Å) | 2.762 | 7.260 | 7.198 |
| b (Å) | 2.762 | 11.110 | 11.166 |
| c (Å) | 3.668 | 2.774 | 11.136 |
| $\beta$ | 90 | 90 | 90 |

Ir

(1a) (000) | (4e)(1$\frac{1}{4}$0) | (4d)(1$\frac{1}{4}$0) |
| (4g)(0.267 0 0) | (4e)(0.269 0 0) | (4g)(0.273 .254 $\frac{1}{4}$) |
| (4g)(0.268 .002 $\frac{1}{4}$) |

V

(1d) (1$\frac{1}{2}$10) | (4j)(0 .364 $\frac{1}{2}$) | (4f)(0 .358 .134) |
| (4j)(0 .364 $\frac{1}{2}$) | (4f)(0 .632 .117) | (4f)(0 .895 .117) |
| (4j)(0 .115 1) | (4f)(0 .123 .132) |

| Volume | 27.97 | 27.97 | 27.97 |
| Energy | 0 | 10.98 | 20.61 |
| Stability | Unstable | Unstable | Unknown |
| Structure | Source | Space Group | Atoms | a (Å) | b (Å) | c (Å) | β | Ir | V | Volume | Energy | Stability |
|-----------|--------|-------------|-------|-------|-------|-------|---|----|---|--------|---------|-----------|
| Ga₃Pt₅ | $S_3(\frac{1}{4} \frac{1}{2} \frac{1}{2})$ | Cmmm | 8 | 7.222 | 7.800 | 3.974 | 90 | (4e)(1/4 1/2 0) | (2a)(0 0 0) | 27.98 | 20.25 | Unstable |
| | L₁₀ phonon | Amm2 | 8 | 2.788 | 7.146 | 11.230 | 90 | (4d)(0.230 0.752) | (2b)(1/2 0 0.352) | 27.97 | 24.15 | Unstable |
| | $\Gamma_3$ Cmmm phonon | Imm2 | 32 | 7.176 | 11.144 | 11.190 | 90 | (4c)(0.223 0.752) | (4d)(0.384 0.625) | 27.96 | 25.12 | Unknown |
| | H₁(\frac{1}{4}00) Amm2 phonon | | | | | | | (4e)(0.268 0.250 0.252) | (4d)(0.380 0.125) | | | |

Table 4: Continuation of Table 3.
| Structure | Cm | Rh<sub>5</sub>V<sub>3</sub> | αIrV |
|-----------|----|-----------------|------|
| Source    | Δ<sub>1</sub>(0<sub>4</sub>0) Amm<sup>2</sup> phonon | Γ<sub>4</sub> Ga<sub>3</sub>Pt<sub>5</sub> phonon | Exp. |
| Space Group | Cm | Amm<sup>2</sup> | Cmmm |
| Atoms     | 32 | 8 | 4 |
| a (Å)     | 13.303 | 4.070 | 5.816 |
| b (Å)     | 11.147 | 7.029 | 6.767 |
| c (Å)     | 6.652 | 7.822 | 2.823 |
| β         | 65.3 | 90 | 90 |
| Ir        | (2a)(.999 0 .504) (2a)(.265 0 .974) (2a)(.485 0.534) (2a)(.755 0 .994) (2a)(.870 0 .256) (2a)(.140 0 .715) (2a)(.360 0 .276) (2a)(.626 0 .745) (2b)(.992 .249 .521) (2b)(.260 .252 .983) (4b)(.365 .748 .267) (4b)(.133 .251 .729) | (4d)(0 .219 .724) (4e)( 0.281 .001) | (4j)(0 .216 1/2) |
| V         | (4b)(.178 .130 .344) (4b)(.175 .380 .362) (4b)(.312 .134 .625) (4b)(.312 .385 .625) (4b)(.064 .125 .117) (4b)(.060 .375 .133) (4b)(.447 .130 .906) (4b)(.450 .380 .888) | (2a)(0 0 .032) (2a)(0 0 .457) (2b)( 1/2 .218) (2b)( 1/2 .793) | (4g)(.297 0 0) |
| Volume    | 27.97 | 27.97 | 27.78 |
| Energy    | 25.67 | 41.14 | 67.25 |
| Stability | Unknown | Metastable | Ground State |
is metastable is the experimentally observed Rh$_5$V$_3$ phase, albeit in a 50-50 composition.

However, none of the unstable structures shows any sign of relaxing into the ground state structure. In fact, looking at Tables 3-5, we see that all of the structures except $\alpha$IrV have approximately the same volume, 27.97 Å$^3$/formula unit in IrV. The ground state $\alpha$IrV structure, on the other hand, has a somewhat smaller volume, 27.78 Å$^3$/formula unit. While this may seem insignificant, the volume change from one phase to another is a signal of a first-order transition, although only weakly first order here.

This paper also shows a mechanism for generating new candidate intermetallic phases: look for vibrationally unstable modes in a high energy structure, construct a supercell which will mimic that mode within the frozen-phonon calculation, and relax the cell. Using this method we found five new structures, as well as one, Rh$_5$V$_3$, which had been seen before but which we had not been aware of until this research started. Many more new structures can undoubtedly be derived from just this system, but further research in this area is currently restricted by the time needed to search the Brillouin zone of a given crystal. For the eight atom supercells computing the phonon spectrum at 45 q-points with a reasonable number ($\leq$ 64) of processors took days to weeks to complete, depending on the symmetry of the crystal. Calculating the phonons for larger unit cells will of course take longer, although we will have fewer q-points to consider, alleviating some of the $O[N^3]$ increase due to the larger unit cell. One of our research goals will be to construct a “set-and-forget” mechanism which will search the Brillouin zone of an initial structure (here L1$_0$) find all unstable modes with supercells containing a given number of atoms, relax those modes, and repeat, until all such structures have been found. Even restricting ourselves to all possible binary intermetallics this will require a Grand Challenge computational program.

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[1] S. Curtarolo, D. Morgan, K. Persson, J. Rodgers, G. Ceder, Predicting crystal structures with data mining of quantum calculations, Phys. Rev. Lett. 91 (13) (2003) 135503. doi:10.1103/PhysRevLett.91.135503
URL http://link.aps.org/abstract/PRL/v91/e135503

[2] J. M. Sanchez, F. Ducastelle, D. Gratias, Generalized cluster description of multicomponent systems, Physica A 128 (1984) 334–350.

[3] D. de Fontaine, Cluster approach to order-disorder transformations in alloys, Solid State Physics 47 (1994) 33–176.

[4] A. Zunger, First-principles statistical mechanics of semiconductor alloys and intermetallic compounds, in: P. E. A. Turchi, A. Gonis (Eds.), Statics and Dynamics of Alloy Phase Transitions, NATO ASI Series, Ser. B, Plenum Press, New York, 1994, pp. 361–419.

[5] A. van de Walle, G. Ceder, Automating first-principles phase diagram calculations, Journal of Phase Equilibria 23 (2002) 348.

[6] D. Lerch, O. Wieckhorst, G. L. W. Hart, R. W. Forcade, S. Müller, Constructing cluster expansions for arbitrary lattices from minimal user input, Model. Simul. Mater. Sci. Eng. 17 (2009) 055003.

[7] M. J. Mehl, D. A. Papaconstantopoulos, Applications of a tight-binding total-energy method for transition and noble metals: Elastic constants, Phys. Rev. B 54 (7) (1996) 4519–4530. doi:10.1103/PhysRevB.54.4519
URL http://link.aps.org/abstract/PRB/v54/p4519

[8] H. Haas, C. Z. Wang, M. Fähnle, C. Elsässer, K. M. Ho, Environment-dependent tight-binding model for molybdenum, Phys. Rev. B 57 (3) (1998) 1461–1470. doi:10.1103/PhysRevB.57.1461
URL http://link.aps.org/abstract/PRB/v57/p1461

[9] M. S. Daw, M. I. Baskes, Embedded-atom method: Derivation and application to impurities, surfaces, and point defects in crystals, Phys. Rev. B 29 (12) (1984) 6443–6453. URL http://link.aps.org/abstract/PRB/v29/p6443

[10] O. Levy, G. L. W. Hart, S. Curtarolo, Uncovering compounds by synergy of cluster expansion and high-throughput methods.
[11] G. L. W. Hart, R. Forcade, Generating derivative structures: Algorithm and applications, Phys. Rev. B 77 (2008) 224115.

[12] G. L. W. Hart, R. Forcade, Generating derivative structures from multilattices: Application to hcp alloys, Phys. Rev. B 80 (2009) 014120.

[13] B. C. Giessen, N. J. Grant, New intermediate phases in transition metal systems. ii J. Less-Common Metals 18 (1965) 1080–1081. doi:10.1107/S0365110X65002566. URL http://dx.doi.org/10.1107/S0365110X65002566

[14] B. C. Giessen, P. N. Dangel, N. J. Grant, New phases in the vanadium-iridium system and a tentative constitution diagram J. Less-Common Metals 13 (1) (1967) 62–70. doi:10.1016/0022-5088(67)90047-1. URL http://dx.doi.org/10.1016/0022-5088(67)90047-1

[15] R. M. Waterstrat, R. C. Manuszewski, The vanadium-rhodium constitution diagram J. Less-Common Metals 52 (2) (1977) 293–305. doi:10.1016/0022-5088(77)90010-8. URL http://dx.doi.org/10.1016/0022-5088(77)90010-8

[16] R. M. Waterstrat, B. Dickens, The crystal structure of v3rh5 J. Less-Common Metals 31 (1) (1973) 61–67. doi:10.1016/0022-5088(73)90130-6. URL http://dx.doi.org/10.1016/0022-5088(73)90130-6

[17] W. B. Pearson, The Crystal Chemistry and Physics of Metals and Alloys, Wiley-Interscience, New York, London, Sydney, Toronto, 1972.

[18] B. Chen, H. F. Franzen, A second-order phase transition in v0.54ir0.46 J. Less-Common Metals 159 (1990) 343–347. doi:10.1016/0022-5088(90)90162-D. URL http://dx.doi.org/10.1016/0022-5088(90)90162-D
[19] H. Baker (Ed.), *Alloy Phase Diagram*, ASM Handbooks Online, ASM International, 1991.
URL http://www.asmmaterials.info/

[20] B. M. Klein, R. E. Cohen, Anharmonicity and the inverse isotope effect in the palladium-hydrogen system, *Phys. Rev. B* 45 (21) (1992) 12405–12414. doi:10.1103/PhysRevB.45.12405
URL http://link.aps.org/abstract/PRB/v45/p12405

[21] S. Baroni, P. Giannozzi, A. Testa, Green’s-function approach to linear response in solids, *Phys. Rev. Lett.* 58 (18) (1987) 1861–1864. doi:10.1103/PhysRevLett.58.1861
URL http://link.aps.org/doi/10.1103/PhysRevLett.58.1861

[22] S. Baroni, S. de Gironcoli, A. D. C. and Paolo Giannozzi, Phonons and related crystal properties from density-functional perturbation theory, *Rev. Mod. Phys.* 73 (2) (2001) 515–562. doi:10.1103/RevModPhys.73.515
URL http://link.aps.org/doi/10.1103/RevModPhys.73.515

[23] X. Gonze, Adiabatic density-functional perturbation theory, *Phys. Rev. A* 52 (2) (1995) 1096–1114. doi:10.1103/PhysRevA.52.1096
URL http://link.aps.org/doi/10.1103/PhysRevA.52.1096

[24] S. Lee, R. Hoffmann, Bcc and fcc transition metals and alloys: A central role for the jahn-teller effect, *J. Am. Chem. Soc.* 124 (17) (2002) 4811–4823. doi:10.1021/ja0114557
URL http://dx.doi.org/10.1021/ja0114557

[25] W. Kohn, L. J. Sham, Self-consistent equations including exchange and correlation effects, *Phys. Rev.* 140 (4A) (1965) A1133–A1138.
URL http://link.aps.org/abstract/PR/v140/pA1133

[26] P. Hohenberg, W. Kohn, Inhomogeneous electron gas, *Phys. Rev.* 136 (3B) (1964) B864–B871.
URL http://link.aps.org/abstract/PR/v136/pB864

[27] J. P. Perdew, K. Burke, M. Ernzerhof, Generalized gradient approximation made simple, *Phys. Rev. Lett.* 77 (18) (1996) 3865–3868.
URL http://link.aps.org/abstract/PRL/v77/p3865
[28] G. Kresse, J. Hafner, *Ab initio* molecular dynamics for open-shell transition metals. Phys. Rev. B 48 (17) (1993) 13115–13118.
URL [http://link.aps.org/abstract/PRB/v48/p13115](http://link.aps.org/abstract/PRB/v48/p13115)

[29] G. Kresse, J. Hafner, *Ab initio* molecular-dynamics simulation of the liquid-metal/amorphous-metal transition. Phys. Rev. B 49 (20) (1994) 14251–14269.
URL [http://link.aps.org/abstract/PRB/v49/p14251](http://link.aps.org/abstract/PRB/v49/p14251)

[30] G. Kresse, Ph.D. thesis, Technische Universität Wien, Vienna (1993).

[31] G. Kresse, D. Joubert, From ultrasoft pseudopotentials to the projector augmented-wave method. Phys. Rev. B 59 (3) (1999) 1758–1775.
doi:10.1103/PhysRevB.59.1758
URL [http://link.aps.org/abstract/PRB/v59/p1758](http://link.aps.org/abstract/PRB/v59/p1758)

[32] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. D. Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Scuserio, A. P. Seitsonen, A. Smogunov, P. Umari, R. M. Wentzcovitch, Quantum espresso: a modular and open-source software project for quantum simulations of materials. J. Phys.: Condens. Matt. 21 (39) (2009) 395502.
doi:10.1088/0953-8984/21/39/395502
URL [http://iopscience.iop.org/0953-8984/21/39/395502/](http://iopscience.iop.org/0953-8984/21/39/395502/)

[33] M. J. Gillan, Calculation of the vacancy formation energy in aluminium. J. Phys.: Condens. Matt. 1 (4) (1989) 689–711.
URL [http://stacks.iop.org/0953-8984/1/689](http://stacks.iop.org/0953-8984/1/689)

[34] H. T. Stokes, D. M. Hatch, B. J. Campbell, Isotropy, [http://stokes.byu.edu/isotropy.html](http://stokes.byu.edu/isotropy.html) (2007).

[35] B. J. Campbell, H. T. Stokes, D. E. Tanner, D. M. Hatch, Isodispalce: a web-based tool for exploring structural distortions, Journal of Applied Crystallography 39 (2006) 607–614.

[36] S. C. Miller, W. F. Love, Tables of irreducible representations of space groups and co-representations of magnetic space groups, Pruett, Boulder, 1967.
[37] C. Kittel, Introduction to Solid State Physics, 7th Edition, John Wiley & Sons, New York, 1996, Ch. 3, pp. 80–95.

[38] M. Born, K. Huang, Dynamical Theory of Crystal Lattices, Oxford at the Clarendon Press, Oxford, 1966, Ch. 11-13.

[39] K. Schubert, S. Bhan, W. Burkhardt, R. Gohle, H. G. Meissner, M. Pötzschke, E. Stolz, *Einige strukturelle Ergebnisse an metallischen Phasen (5)*, Naturwissenschaften 47 (13) (1960) 303. doi:10.1007/BF00600960. URL http://dx.doi.org/10.1007/BF00600960

[40] P. Villars, L. D. Calvert (Eds.), Pearson’s Handbook of Crystallographic Data for Intermetallic Phases, 2nd Edition, Vol. IV, ASM International, Materials Park, Ohio, 1991, p. 5091.