Rangel-Vazquez, Norma-Aurea; Sánchez-López, Carlos; Rodríguez Felix, Francisco
Spectroscopy Analyses of Polyurethane/Polyaniline IPN Using Computational Simulation (Amber, MM+ and PM3 Method)
Polímeros: Ciência e Tecnologia, vol. 24, núm. 4, julho-agosto, 2014, pp. 453-463
Associação Brasileira de Polímeros
São Paulo, Brasil

Available in: http://www.redalyc.org/articulo.oa?id=47032470008
Spectroscopy Analyses of Polyurethane/Polyaniline IPN Using Computational Simulation (Amber, MM+ and PM3 Method)

Norma-Aurea Rangel-Vazquez, Carlos Sánchez-López
Departamento de Metal Mecánica, Instituto Tecnológico de Aguascalientes, México

Francisco Rodríguez Felix
Departamento de Investigación y Posgrado en Alimentos, Universidad de Sonora, México

Abstract: The polyurethane (PU)/polyaniline (PANI) IPN, at three composition ratios (95/05, 75/25, 60/40), were analyzed, after the action of the urease enzyme, by means of the Hyperchem 8v software. Theoretical analysis of Gibbs free energy, FTIR, and 13C-NMR were done. The generation of functional groups due to hydrolysis was analyzed by FTIR. The 13C-NMR spectra of the three ratios showed a superposition of the individual spectra of PU and PANI and an appreciable decrease of the signals due to the action of the urease enzyme. The orbits showed that the electron density of the HOMO (Highest Occupied Molecular Orbital) is predominantly localized on the C-C bond, indicating that the binding of PU/PANI with the urease enzyme contains a vacant d-orbital. It is concluded that the adsorption of the urease enzyme increases the degree of hydrolysis of the IPN.

Keywords: PU, PANI, Hyperchem 8v.

Introduction

Polymers are known to have good insulating properties and are among the most used materials in the modern world. However, it has been discovered that there are some polymers which have conducting properties[1-3]. There are at least four major classes of semiconducting polymers that have been developed so far. They include conjugated conducting polymers, charge transfer polymers, ionic conducting polymers and conductive filled polymers[4,5]. The potential uses for conductive filled polymers have since been multiplied due to their ease of processing, good environmental stability and wide range of electrical properties[6].

Conducting polymers are polymer materials with metallic and semiconductor characteristics, a combination of properties not exhibited by any other known material. A principal property of a conductive polymer is the presence of conjugated double bonds along the backbone of the polymer. In conjugation, the bonds between the carbon atoms are alternating single and double. Since the electrons in a conjugated system are only loosely bound, electron flow may be possible.

Every bond contains a localized “σ” bond which forms a strong chemical bond. In addition, every double bond also contains a less strongly localized “π” bond which is weaker. These enable the electrons to be delocalized over the whole system, and then be shared by many atoms. This means that this kind of electrons may move around the whole system. However, conjugation is not enough to make the polymer material conductive[7-10].

In addition, the polymer material needs to be doped for electron flow to occur. Doping is either the addition of electrons (reduction reaction) or the removal of electrons (oxidation reaction) from the polymer. An oxidation doping (removal of electrons) can be done using iodine. The iodine attracts an electron from the polymer from one of the bonds. Once doping has occurred, the electrons in the bonds are able to “jump” around the polymer chain. As the electrons are moving along the molecule, electric current occurs. For better conductivity the molecules must be well ordered and closely packed to limit the distance “jumped” by the electrons. The conductivity of conducting polymers can be tuned by chemical manipulation of the polymer backbone, by the nature of the dopant, by the degree of doping, and by blending with other polymers[1,2,11].

PANI is a kind of intrinsically conducting polymer consisting of phenylenediamine (PPD) and quinonediimine structure as shown in Figure 1[12-16]. The morphology of PANI, granular or tubular, depends on the acidity conditions during the reaction rather than on the chemical structure of the acid[17]. Like many other conducting polymers, PANI is not soluble in common organic solvents because of the stiffness of its backbone and hydrogen bonding interaction between adjacent chains. Improving the processability of PANI, has made possible its use in many applications as electrodes, biosensors, batteries, anti-static coatings, gas sensors, membranes and light emitting diodes[12,18-22].

PANI has resonance structures that resemble the aromatic or quinoid forms. In its neutral state the polymer is not conductive, they become conductive only in the oxidized state and have a conductivity of 104 (Ω cm)−1 for PANI, respectively. The charge associated with the oxidized state is typically delocalized over several units of this polymer and can form a radical cation (polaron) or a dication (bipolaron). The introduction of charge to the chain can transform the aromatic structure in to the lower band gap quinoid structure[23].
The majority of polymers selected to prepare PANI composites is insulating polymeric materials with excellent physical properties, such as rubber, plastic and textile. Many papers dealing with the preparation of the rubber–PANI, plastic–PANI, and others have been published. The reason why PU is chosen is due to its balanced flexibility and hardness, good film forming property and resistance to solvent. Moreover, the interaction between NH in PANI and NHCOO in PU may be a bonus to improve the compatibility of the system.

The strong tendency of PANI and PU to form hydrogen bonding has been verified by Luo et al. and Rangel-Vázquez et al. through different spectroscopic techniques, which was believed to be responsible for a certain degree of phase mixing in PANI/PU blend. In the characterization of PANI filled PU/PMMA interpenetrating network system, Luo et al. and Wang et al. mentioned that the hydrogen bonding formation between NH of PANI and NHCOO group of PU/PMMA polymer network had some effect on the physical, mechanical, electrical as well as thermal properties of the conductive IPN, also suggested that the interaction between the carboxyl groups in PU and imine groups in PANI could induce miscibility of PANI/PU blends. The combination of PANI segments with blocks of another polymer has been recently reported, such as the grafting of polyethylene glycol or polyacrylic acid onto a PANI backbone, and a block copolymer consisting of PU and PANI oligomer, etc. The use of a thermoplastic elastomer with conducting polymers is very attractive due to the combination of mechanical properties and a processability which does not require vulcanization.

Enzymes exhibit a number of features that make them as advantageous as compared to conventional chemical catalysts. Foremost among them are a high level of catalytic efficiency, often far superior to chemical catalysts, and a high degree of specificity that allows them to discriminate not only between reactions but also between substrates (substrate specificity), similar parts of molecules (regiospecificity) and between optical isomers (stereospecificity). The enzymatic hydrolysis is a heterogeneous process that is affected by the mode of interaction between the enzymes and the polymeric chains and involves typically four steps: (1) diffusion of the enzyme from the bulk solution to the solid surface, (2) adsorption of the enzyme on the substrate, resulting in the formation of the enzyme–substrate complex, (3) catalysis of the hydrolysis reaction, and (4) diffusion of the soluble degradation products from the solid substrate to the solution.

The rate of the global reaction is controlled by the slowest step. The enzymatic hydrolysis of solid substrates are normally characterized by an enzyme saturation point; in this enzyme concentration, no further increase in the degradation rate is observed when more enzymes are added. This has been attributed to a decrease in the exposed polymer surface as the enzyme molecules saturate the surface, and appears to be limiting the progress of degradation of some biomaterials. Quantum mechanical calculations have extensively been used to study the energetics of simple chemical reactions and to understand the electrostatic interactions between compounds. Their application to polymeric systems, until very recently was not available. This was mainly due to very large numbers of atoms in polymeric structures requiring powerful computers and very long computer times. With advances in both computer and software technology, now it is possible to do quantum mechanical calculations or precise simulations on polymeric systems. In this work, the computational analysis was applied to PU, PANI and PU/PANI (3 ratios), where Gibbs free energy, FTIR, 13C-NMR, and molecular orbitals were calculated after the action of the urease enzyme.

**Experimental Section**

**Methods**

Since sketching often creates the structure in a high energy configuration, and starting a simulation from such an unoptimized structure can lead to erroneous simulation results, after it has been sketched, it is necessary to carry out the geometry optimization or energy minimization of the system being examined. In this work three aspects were considered for selection of geometry optimization method and algorithm: system size, convergence level and force field used. Next a minimization algorithm is chosen to find the potential energy minimum corresponding to the lowest energy structure. Iteration number and convergence level lead optimal structure.

The geometry optimization (Amber, MM+ and PM3 method, respectively) was calculated by means of the Polak-Ribiere algorithm with a gradient of 0.001 Kcal/(Å mol) in vacuum, 1 spin multiplicity, convergence limit of 0.001, and 1000 iteration limits. Table 1 shows the urease enzyme properties.

### Table 1. Properties of urease enzyme.

| Name         | Enzyme          | Source  | Reaction temperature | pH | Mw     | Purity |
|--------------|-----------------|---------|----------------------|----|--------|--------|
| Urease       | Hydrolase       | Bean    | 25 °C                | 8  | 162,000 g/mol | 80%    |
FTIR

The infrared spectrum is commonly obtained by passing infrared electromagnetic radiation through a sample that possesses a permanent or induced dipole moment and determining what fraction of the incident radiation is absorbed at a particular energy. In an absorption spectrum, the energy of each peak corresponds to the frequency of the vibration of a molecule part, thus allowing qualitative identification of certain bond types in the sample. The FTIR was obtained by first selecting menu compute, vibrational and rotational options. Once completed this analysis, using the vibrational option, the FTIR spectrum pattern is obtained for two methods of analysis[31]. Tables 2 and 3 show the experimental values of PU and PANI, respectively.

| Assignment                  | Wavenumber (cm⁻¹) |
|-----------------------------|-------------------|
| NH, OH stretching           | 3520 ± 3350       |
| CH asymmetric stretching    | 2936 ± 2919       |
| C=O stretching              | 1724 ± 1715       |
| CN and NH stretching        | 1616              |
| NH (Amide II) bending       | 1525 ± 1550       |
| C–O–C                      | 1255              |
| C-N, C-C and C-O stretching| 1187 ± 1015       |

Table 3. FTIR experimental values of PANI[14,22].

| Assignment                  | Wavenumber (cm⁻¹) |
|-----------------------------|-------------------|
| N-H stretching              | 3450              |
| C–C ring asymmetric and symmetric stretching vibrations | 1575 and 1519 |
| Benzene structure           | 1485              |
| N–H and C–C bending vibrations | 1325 and 1291     |
| C-N stretching of a secondary aromatic amine | 1310 |
| C-H bending (in-plane and out-of-plane) | 1138 and 834     |

NMR analysis

13C-NMR analyses were obtained by means of TNDO/2 method to determine the shielding and coupling constants of different IPN’s. The electrostatic potential property was obtained with 3D mapped isosurface representation. Tables 4 and 5 show the experimental values of PU and PANI, respectively.

| Assignment                  | PPM   |
|-----------------------------|-------|
| C(O)ONH                     | 159, 154 |
| C=O                        | 137 – 129, 125 |
| C=C                        | 121, 119, 100 |
| CH₂O                       | 72, 66  |
| C – C                      | 40     |
| CH₃                        | 33 – 22, 20 |
| CH₄                        | 14     |
| NH group                   | 8      |
| CH₂                        | 4.2, 1.8 |

2.4 Electrostatic potential

After obtaining a Gibbs free energy or geometry optimization using different methods, we can plot two-dimensional contour diagrams of the electrostatic potential surrounding a molecule, the total electronic density, the spin density, one or more molecular orbitals, and the electron densities of individual orbitals. When you select the appropriate option, in the Contour plot dialog box, Hyperchem software displays the electrostatic potential as a contour plot. Choose the values for the starting contour and the contour increment so that you can observe the minimum (typically about –0.5 for polar organic molecules) and so that the zero potential line appears.

A menu plot molecular graph, the electrostatic potential property is selected and then the 3D representation mapped isosurface for both methods of analysis. Atomic charges indicate where large negative values (sites for electrophilic attack) are likely to occur. However, the largest negative value of the electrostatic potential is not necessarily adjacent to the atom with the largest negative charge[31].

2.2 Quantum chemical calculations

To understand the hydrolysis of different PU/PANI ratios, after hydrolysis reaction with urease enzyme, the molecular electronic structure calculations have been performed. These calculations yielded HOMO and LUMO (Lowest Unoccupied Molecular Orbital) energies. Among quantum chemical methods, the Density functional theory (DFT) is found to be a suitable method for our calculations.

All the calculations were performed using Amber, MM+ and PM3 method. Although the three semi-empirical parameters are optimized primarily to reproduce thermo-chemistry of small organic molecules, it has been proven to perform exceptionally well for relatively large organic molecules of heteroatoms.

Results and Discussions

The geometry optimization was used to locate a minimum of energy, on the surface, of the potential energy surface, and get the balance structure of a molecular system. Using the Hyperchem 8y software, by means of the Amber, MM+ and PM3 method, with the Polak-Ribiere algorithm was possible calculate the Gibbs free energy (ΔG) of PU and PANI (see Table 6), where
were observed the energy for diverse models, proven that the reaction was carried out.

Table 7 shows the $\Delta G$ that was used to locate a minimum of energy, in the surface, of the potential energy surface, and get the balance structure of a molecular system, using the Hyperchem 8v software by means of the semi-empirical methods (Amber, MM+ and PM3, respectively) and the Polak-Ribiere algorithm was possible calculate the $\Delta G$ of different IPN’s. The negative values obtained shows that the reaction is spontaneous and the reaction mechanisms were adjusted to first order in each one of the relationships (95/05, 75/25 and 50/50, respectively). These results are similar to those reported by Rangel (2007) who determined that the blended and supported PU/HA composites are vulnerable to the hydrolysis reactions by means of the action of the urease enzyme, which it was carried out by means of hydrophobic adsorption in the surface followed by the hydrolysis of the ester bond of the PU, being proven the kinetics of first order[29]. Ayad et al (2003) studied the kinetics formation of PU/PANI, concluding that the reaction was adjusted in first order[41]. Ma et al (2008) determined the kinetics of polymerization of the aniline of the first order, concluding that, the increment of the concentration of radicals generated that polymerization began[40].

**FTIR analyses**

The strong tendency of PANI and PU to form hydrogen bonding has been verified by Rodrigues et al. through various spectroscopic techniques, which was believed to be responsible for a certain degree of phase mixing in PANI/PU blend[43]. Ma et al (2009) also confirmed the hydrogen bonding between PANI and PU, and the strong hydrogen bonding in the blend contributed to better miscibility, higher conductivity and tensile strength[44].

| Method | $\Delta G$ (Kcal/mol) | n | $\Delta G$ (Kcal/mol) | n |
|--------|----------------------|---|----------------------|---|
| Amber  | – 4583               | 0.993          | – 3914               | 0.997         |
| MM+    | – 4207               | 0.981          | – 3205               | 0.990         |
| PM3    | – 3056               | 0.974          | – 2885               | 0.984         |

| Method | $\Delta G$ (Kcal/mol) | n | $\Delta G$ (Kcal/mol) | n |
|--------|----------------------|---|----------------------|---|
| Amber  | 0.999                | – 43254      | 0.994                | – 40192       |
| MM+    | 24726                | 0.997        | – 21879              | 0.991         |
| PM3    | – 26304              | 0.996        | – 25081              | 0.987         |

Table 7. $DG$ and reaction order (n) of PU/PANI respectively.

| VIBRATIONAL MODE | EXPERIMENTAL VALUES (cm$^{-1}$) | THEORETICAL VALUES (cm$^{-1}$) |
|------------------|--------------------------------|-------------------------------|
| Amine secondary  | 3430, 3280                     | 3553                          | 3551            | 3548            |
| CH vibration     | 2618                           | 2559                          | 2956            | 2955            |
| C=C double (quinoid rings) | 1593                 | 1675                          | 1674            | 1675            |
| CC stretching    | 1612                           | 1638                          | 1622            | 1619            |
| Stretching vibration for benzoid and quinoid | 1573, 1500                      | 1585                          | 1537            | 1465            |
| Ph-NH            | 1138                           | 1216                          | 1213            | 1211            |
| CN               | 1174                           | 1157                          | 1149            | 1143            |
| CH               | 1201                           | 1205                          | 1209            | 1205            |
| Vibration for benzoid and quinoid | ---------             | 953                           | 947             | 931             |

Table 6. Gibbs free energy ($\Delta G$) and reaction order (n) for PANI and PU respectively.

Table 8. FTIR results of PANI.
ring CC is stretching partially mixed with CH bending. At 953 and 931 cm\(^{-1}\) correspond to the vibrations of the quinoid and benzoid ring\(^{[48]}\).

IR spectroscopy is an important experimental technique, which is widely used to determine the extent of hydrogen bonding in polymeric systems and especially in PUs. By determining the shifts in peak positions, a semi-quantitative measure of the hydrogen bond strength can be obtained from IR spectra. Quantum mechanical calculations are also very useful in constructing the IR spectrum of single molecules or complex systems, where peak shifts and related energy changes due to hydrogen bonding can quantitatively be obtained\(^{[30]}\). In bands corresponding to the urethane group (-CH\(_2\), 2867-2825 cm\(^{-1}\)), two shoulder peaks appears at 1838-1835 cm\(^{-1}\), which is assigned to the carbonyl group (C=O) can be observed in Table 9, the peaks between 1698 and 1697 cm\(^{-1}\) corresponds to C=C stretching peak in the benzene rings. At 1489-1482 cm\(^{-1}\) were assigned to stretching of benzene ring. The peaks from 1142 to 1138 cm\(^{-1}\) because of vibration of CH in benzene ring and stretching of CN band have been observed\(^{[28]}\).

Tables 10 to 12 show FTIR spectra of 3 PU/PANI ratios, where, the characteristic bands of the hydrolysis were appreciated. From 1412 to 1370 cm\(^{-1}\) are attributed to the stretching C-OR, and from 1270 to 1200 cm\(^{-1}\) were assigned to the stretching of the bond C-OR of the urethane group formed by the hydrolysis\(^{[13,47]}\). Finally, it was appreciated that the degradation was made mainly on

| Table 9. FTIR results of PU. |
|-------------------------------|
| **VIBRATIONAL MODE** | **EXPERIMENTAL VALUES** (cm\(^{-1}\)) | **THEORETICAL VALUES**  
| Amber | MM+ | PM3 |
|------|------|------|
| CH\(_2\) | 2912, 2807 | 2867 | 2825 | 2825 |
| C=O | 1582 | 1698 | 1697 | 1697 |
| C=O | 1720 | 1838 | 1835 | 1835 |
| CN | 1375 | 1489 | 1482 | 1482 |
| Benzene ring | 1231 | 1290 | 1289 | 1289 |
| CN | 1210 | 1276 | 1274 | 1274 |
| CH | 1115 | 1142 | 1138 | 1138 |

| Table 10. FTIR theoretical analysis results (Amber method) for PU/PANI relations modified by urease enzyme. |
|-----------------------------------------------------|
| **VIBRATIONAL MODE** | **95/05** (cm\(^{-1}\)) | **75/25** (cm\(^{-1}\)) | **50/50** (cm\(^{-1}\)) |
|-----------------|-----------------|-----------------|-----------------|
| C=O symmetric stretch of urethane group (hydrolysis) | 1270 | 1258 | 1247 |
| NH (ternary amide) | 1220 | 1215 | 1201 |
| CN (urethane group) | 1260 | 1257 | 1254 |
| C=O symmetric stretch (urethane group) | 1412 | 1406 | 1381 |
| Benzene ring (urea) | 1496 | 1435 | 1431 |
| Amide I (urea) | 1690 | 1687 | 1680 |
| C=O carbonyl stretching vibration (intensity less) | 1765 | 1754 | 1749 |

| Table 11. FTIR theoretical analysis results (MM+ method) for PU/PANI relations modified by urease enzyme. |
|-----------------------------------------------------|
| **VIBRATIONAL MODE** | **95/05** (cm\(^{-1}\)) | **75/25** (cm\(^{-1}\)) | **50/50** (cm\(^{-1}\)) |
|-----------------|-----------------|-----------------|-----------------|
| C=O symmetric stretch of urethane group (hydrolysis) | 1263 | 1254 | 1228 |
| NH (ternary amide) | 1216 | 1207 | 1198 |
| CN (urethane group) | 1258 | 1257 | 1252 |
| C=O symmetric stretch (urethane group) | 1410 | 1402 | 1372 |
| Benzene ring (urea) | 1475 | 1466 | 1449 |
| Amide I (urea) | 1685 | 1684 | 1679 |
| C=O carbonyl stretching vibration (intensity less) | 1766 | 1760 | 1745 |
the relationship 95/05 due to a bigger presence of PU in the IPN’s.

Also, from 1765 to 1749 cm\(^{-1}\) were attributed to the decreased of the stretching C=O, as well as in the bands related with the groups amide I (urea) and the benzene ring (urea), whose bands were determined from 1690 to 1680 and from 1496 to 1431 cm\(^{-1}\), respectively. From 1220 to 1201 cm\(^{-1}\) was assigned to the amide III of the urethane group, at 1220 cm\(^{-1}\) corresponded to the NH and CN of stretching (ternary amide) of the urea and urethane group.

**NMR analyses**

Figure 2 shows \(^{13}\)C-NMR spectra with 3 ratios (95/05, 75/25 and 50/50, respectively), after attack with urease enzyme, where appears merely as a superposition of the individual spectra of the polymers (PANI and PU). From 18 to 25 ppm are attributed to the CH bonds of PANI which increased with the PANI percentage in the IPNs, also, from 180 ppm diverse signs arose as a result of the reaction between the isocyanate (HDI) with the hydroxyl groups of 1,6-hexanediol and isopropanol. Two strong and sharp peaks centered at 48 and 75 ppm are ascribed to the soft-segment carbons. The peaks at 200 ppm were attributed to the resonance of carbonyl (urethane). The resonance at 40 ppm was assigned to –CH\(_2\) – groups of HDI. According to the results of Salis et al., the resonance of –CH\(_2\) – in HDI is restricted by the neighboring PANI oligomers\(^{46}\).

In addition, the aromatic region ranging from 140 to 155 ppm has contributions from all of the PANI carbons. However, because the chemical shifts of resonance peaks in this region are affected by the disturbance of electron clouds from individual components, the peaks presenting in the spectrum are broader as compared to those of neat PUs. In this region, the peak at 120 ppm are assigned to the benzene protonated ring carbons of PANI, while the peak at 145 ppm is associated with the quinoid protonated ring carbons of PANI\(^{28}\).

Table 12. FTIR theoretical analysis results (PM3 method) for PU/PANI relations modified by urease enzyme.

| VIBRATIONAL MODE                                      | 95/05 (cm\(^{-1}\)) | 75/25 (cm\(^{-1}\)) | 50/50 (cm\(^{-1}\)) |
|-------------------------------------------------------|----------------------|----------------------|----------------------|
| C-O symmetric stretch of urethane group (hydrolysis) | 1252                 | 1231                 | 1200                 |
| NH (ternary amide)                                    | 1213                 | 1204                 | 1195                 |
| CN (urethane group)                                   | 1257                 | 1253                 | 1249                 |
| C-O symmetric stretch (urethane group)                | 1405                 | 1400                 | 1370                 |
| Benzene ring (urea)                                   | 1460                 | 1429                 | 1432                 |
| Amide I (urea)                                        | 1684                 | 1683                 | 1683                 |
| C=O carbonyl stretching vibration (intensity less)     | 1762                 | 1760                 | 1744                 |

**Figure 2.** \(^{13}\)C -RMN spectra where (a) PU and (b) PANI, using AMBER method.
PANI and the protonated HDI carbons, while the peak at 225 ppm is associated with the quinoid protonated ring carbons of PANI. Some peaks in the regions of aromatic and urethane carbonyl are almost invisible, which is also ascribed to the electron clouds disturbance arising from the trapped PU-PANI. As compared to that of PU-PANI, weaker peaks in the aromatic region are observed, while the peak at 40 ppm shows a little stronger. It is clear that this difference in peak intensity result by means to urease enzyme action\cite{25}.

Electrostatic potential

Figure 4a show the electrostatic potential for 95/05 (PU/PANI) ratio, where is appreciated that the electrostatic potential is from 1.247 (maximum value) to 0.065 (minimum value) due to the electronegative present of the PANI, while in the Figure 4b it is appreciated that the potential diminishes from 0.109 (maximum value) to 0.024 (minimum value), due to the action of the urease enzyme that modify the chemical structure of the molecules generating a loss of the area of interaction of the IPN\cite{42,52,53}. Olivucci et al. also confirmed the hydrogen bonding between PANI and PU, and the strong hydrogen bonding in the blend contributed to better miscibility, higher conductivity and tensile strength\cite{54}.

Molecular orbitals

Figure 5 shows the frontier molecular orbital (highest occupied molecular orbital, HOMO) density distributions of (a) PU and (b) PANI. It is found that the electron density of HOMO for the PU unit is localized mainly on two benzene with small contributions from disocyanate\cite{44}, the preferred configuration of PANI includes an alternate tilt of one aromatic ring with respect to the next (the arene-like rings are not coplanar). In fact, we observe light polarization dependence of PANI films with thicknesses less than 50 Å. Because the HOMO, HOMO-1, and HOMO-2 are enhanced with s-polarized light, though observed in both p and s-polarized light, it is clear that the benzene rings (and the chain length) are mostly parallel with the surface\cite{55,56}.

Table 13 shows that for the PU/PANI with urease enzyme, the electron density of HOMO is localized predominantly on the C-C bond. This indicates that the bonding of PU/PANI with enzyme containing d-orbital vacant will be more facile when compared to bonding though benzene ring in the absence of urease enzyme. Our calculations reveal that the enzyme adsorption further increases with the increasing hydrolysis degree and hence the inhibition efficiency. To understand the effect of hydrolysis degree on inhibition efficiency, we have computed the HOMO–LUMO energy gap. Often, EHOMO refers to the tendency of inhibitor to donate electrons Higher value of EHOMO indicates a greater electron donation tendency and, hence, stronger bond formation. On the other hand, lower value of ELUMO indicates more probable electron acceptance of the molecule. This will also be strengthening bond by accepting back donated electrons. The energy gap $\Delta E = ELUMO - EHOMO$ is thereby an important parameter for determining reactivity of the urease molecules towards the adsorption on the PU/PANI surface.
The reactivity of the urease molecule increases with decreasing $\Delta E$, promoting the greater inhibition efficiency of the molecule. Therefore, the reactivity of the urease molecule for adsorption on the PU/PANI surface increases with increasing degree of hydrolysis. Thus, the electronic structure calculations envisage the orbital density distribution of HOMO of the PU/PANI molecule with and without the urease enzyme and predict the electron donating capability of ionomer with varying degree of hydrolysis. Our theoretical findings of an increase in inhibition efficiency with increasing degree of hydrolysis corroborate the experimental observations as mentioned earlier[47].

Figure 4. Electrostatic potential, in where (a) PU/PANI with 95/05 ratio, (b) PU/PANI after attack (15 days) with urease enzyme using Amber method.
Figure 5. Shows the frontier molecular orbital (highest occupied molecular orbital, HOMO) density distributions of (a) PU and (b) PANI.

Table 13. HOMO and LUMO orbitals value for PANI, PU and PU/PANI with urease enzyme.

| ORBITAL | HOMO PANI (eV) | HOMO PU (eV) | HOMO PU/PANI (eV) | LUMO PANI (eV) | LUMO PU (eV) | LUMO PU/PANI (eV) |
|---------|---------------|--------------|-----------------|----------------|-------------|-----------------|
| 1       | −73.320       | −10.75       | 1.670           | −47.71         | −5.75       | 1.790           |
| 5       | −87.590       | −13.22       | 1.500           | −08.38         | −2.87       | 1.850           |
| 50      | −337.82       | −29.55       | −11.69          | 223.11         | −2.87       | 2.820           |
| 100     | −540.04       | ND           | −39.61          | 380.47         | ND          | 21.34           |
Conclusions

Samples of PU/PANI with different composition were analyzed. Using the Hyperchem software 8v by means of the Amber, MM+ and PM3 method, with the Polak-Ribiére algorithm was possible calculate the Gibbs free energy (AG) of PU, PANI and PU/PANI (3 ratios).

FTIR tables showed the characteristic functional groups and different signs by means hydrolysis of PU, PANI and 3 PU/PANI ratios with 3 methods of computing modeling. From 1270 to 1200 cm⁻¹ were assigned to the stretching of the bond C-OR of the urethane group formed by the hydrolysis, in fact, can be appreciated that urease enzyme attacks more strongly to 95/05 due to a bigger presence of PU in the IPN’s.

¹³C-NMR spectra with 3 ratios (95/05, 75/25 and 50/50, respectively), show that appears merely as a superposition of the individual spectra of the polymers (PANI and PU). It is clear that this difference in peak intensity result by means to urease enzyme. The absorption peak at 334 nm is attributed to the transition (LUMO) which is related to the (HOMO) to the lowest unoccupied molecular orbital (LUMO) which is related to π—π* electronic transition. Orbital analysis demonstrated that the electron density of HOMO is localized predominantly on the C-C bond. This indicates that the bonding of PU/PANI with enzyme containing d-orbital vacant. Our calculations reveal that the enzyme adsorption further increases with increasing degree of IPN’s hydrolysis.

References

1. Hesketh, P. J. & Misra, D. - Electrochem. Soc. Interface, 21, p.61 (2012).
2. Arshak, K.; Velusamy, V.; Korostynska, O.; Oliwa-Stasiak, K. & Adley, C. - IEEE Sens. J, 9, p.1942, (2009). http://dx.doi.org/10.1109/JSEN.2009.2032052
3. Angelopoulos, M. - IBM J. Res. Dev., 45, p.57 (2001).
4. Alam, M.; Ansari, A. A.; Shaik, M. R. & Alandin, N. M. - Arabina J. Chem., 6, p.341 (2013). http://dx.doi.org/10.1016/j.arabjc.2012.04.021
5. Pant, H. C.; Patra, M. K.; Negi, S. C.; Bhatia, A.; Vadera, S. R. & Kumar, N. - Bull. Mater. Sci., 29, p.379 (2006). http://dx.doi.org/10.1007/BF02704139
6. MacDiarmid, A. G. - Angew. Chem. Int. Ed., 40, p.2581 (2001). http://dx.doi.org/10.1002/1521-3773(20010716)40:14<2581::AID-Anie2581>3.0.CO;2-2
7. Dai, L. & Mau, A. W. H. - Adv. Mater., 13, p.899 (2001). http://dx.doi.org/10.1002/1521-4095(200107)13:12<899::AID-ADMA899>3.0.CO;2-G
8. Meier, H.; Lehmann, M. & Kolb, U. - Chem. Eur. J., 6, p.2462 (2000). http://dx.doi.org/10.1002/1521-3765(20000703)6:13<2462::AID-CHEM2462>3.0.CO;2-A
9. Bredas, J. L. & Street, G. B. - Acc. Chem. Res., 18, p.309 (1985). http://dx.doi.org/10.1021/ar00118a005
10. Chakane, P. S.; Jain, L. S. & Bhoraskar, S. V. - Trans. SAEST, I, p.35 (2002).
11. Patil, O.; Heeger, A. J. & Wudl, F. - Chem. Rev., 88, p.183 (1988). http://dx.doi.org/10.1021/cr00083a009
12. Luo, J.; Wang, X.; Li, J.; Zhao, X. & Wang, F. - Polymer, 48, p.4368 (2007). http://dx.doi.org/10.1016/j.polymer.2007.05.062
13. Trchová, M.; Sedenkova, M. I.; Tobolkova, E. & Stejskal, J. - Polym. Degrad. Stab., 86, p.179 (2004). http://dx.doi.org/10.1016/j.polymdegradstab.2004.04.011
14. Stejskal, J. & Gilbert, R. G. - Pure Appl. Chem., 74, p.857 (2002).
15. Reza, N. M.; Golbabaei, M.; Moghadam, A. B.; Dinarvand, R. & Sedghi, R. - Int. J. Electrochem. Soc., 3, p.1117 (2008).
16. Del Castillo-Castro, T.; Castillo-Ortega, M. M.; Villarreal, I.; Brown, F.; Grijalva, H.; Pérez-Tello, M.; Nuño-Dolucas, S. M. & Puig, J. E. - Composites Part A, 38, p.639 (2007). http://dx.doi.org/10.1016/j.compositesa.2006.02.001
17. Sedenkova, I.; Trchová, M. & Stejskal, J. - Polym. Degrad. Stab., 93, p.2147 (2008).
18. Zun-Li, M.; Zhong-Li, Z.; Hong, C.; Gui-Ping, N. & Hua-Feng, S. - Carbohydr. Polym., 75, p.660 (2009).
19. Lee, H. Y.; Rwei, S. P.; Wang, L. & Chen, P. H. - Mater. Chem. Phys., 112, p.805 (2008). http://dx.doi.org/10.1016/j.matchemphys.2008.06.050
20. Rangel-Vázquez, N. A.; Salgado-Delgado, R.; García-Hernández, E.; Mendoza-Martínez, A. M. & Gómez-Jimenez, M. A. – Braz. J. Mater. Sci. Eng., 4, p.45 (2010).
21. Zhou, C.; Han, J.; Song, G. & Guo, R. - Eur. Polym. J., 44, p.2850 (2008). http://dx.doi.org/10.1016/j.eurpolymj.2008.01.025
22. Islam, S.; Lakshmi, G. B. V. S.; Siddiqui, A. M.; Husain, M. & Zulfiquar, M. - Int. J. Polym. Sci., 2013, p.1 (2013). http://dx.doi.org/10.1155/2013/307525
23. Abdulla H. S. & Abbo, A. I. - Int. J. Electrochem. Sci., 7, p.10666 (2012).
24. Schmidt, V.; Domenech, S. C.; Soldi, M. S.; Pinheiro, E. A. & Soldi, V. - Polym. Degrad. Stab., 83, p.519 (2004). http://dx.doi.org/10.1016/j.polymdegradstab.2003.08.011
25. Rangel-Vázquez, N. A.; Salgado-Delgado, R.; García-Hernández, E. & Mendoza-Martínez, A. M. - J. Mex. Chem. Soc., 53, p.248 (2009).
26. Wang, T. L.; Yang, C. H.; Shieh, Y. T. & Yeh, A. C. - Eur. Polym. J., 45, p.387 (2009). http://dx.doi.org/10.1016/j.eurpolymj.2008.11.020
27. Vicentini, D. S.; Barra, G. M. O.; Bertolino, J. R. & Pires, A. T. N. - Eur. Polym. J., 43, p.4565 (2007). http://dx.doi.org/10.1016/j.eurpolymj.2007.06.046
28. Rangel-Vázquez, N. A.; Salgado-Delgado, R.; García-Hernández, E. & Mendoza-Martínez, A. M. - Rev. Iber. Polímeros, 11, p.179 (2010).
29. Rangel-Vázquez N. A. - “Políuretano puroso/ Hidroxiapatita para Aplicaciones Biomédicas: Síntesis y Caracterización”, PhD thesis, Instituto Tecnológico de Ciudad Madero, México (2007).
30. Yilgör, E.; Yilgör, I. & Yurtsever, E. - Polymer, 43, p.6551 (2002). http://dx.doi.org/10.1016/S0032-8831(02)00567-0
31. Rangel-Vázquez, N. A. & Rodríguez-Feliz, F. - “Computational Chemistry Applied in the Analyses of Chitosan/Polyvinylpyrrolidone/Mimoso Tenuiflore”, Science Publishing Group, Hong Kong (2013). PMid:24216118.
32. Rangel, N. A.; De Alva, H. E.; Romero, J.; Rivera, J. L.; Álvarez, A. & García, E. - Rev. Iber. Polímeros, 8, p.99 (2007).
33. Pretsch, T.; Jakob, I. & Müller, W. - Polym. Degrad. Stabil., 94, p.61 (2009). http://dx.doi.org/10.1016/j.polymdegradstab.2008.10.012
34. Ngoc, L. P. & Van, B. L. - J Sci Nat, Sci, & Tech, XXIII, p.19 (2007).
35. Wong, C. S. & Badri, K. H. - Mater. Sci. Appl., 3, p.78 (2012).
36. Omed, G. H.; Gelas, A.; Jamal, M.; Tahir, D. A. & Saeed, S. R. - IJAPM, 1, p.101 (2011).
37. Zujovic, Z. D.; Gizdavic-Nikolaidis, M. & Bowmaker, G. A. - Chem. New Zealand, 77, p.87 (2013).
38. Wu, C. S. - Express Polym. Lett., 6, p.465 (2012). http://dx.doi.org/10.3144/expresspolymlett.2012.49
39. Kim, S.; Shin, S.; Spinks, G.; Kim, I. & Kim, S. - J. Appl. Polym. Sci., 96, p.867 (2005). http://dx.doi.org/10.1002/app.21524
40. Araoyinbo, A. O.; Rahmat, A.; Derman, M. Z. & Ahmad, K. R. - Adv. Mater. Lett., 4, p.899 (2013).
41. Ayad, M. M.; Gemae, A. H.; Salahuddin, N. & Shenashin, M. A. - J. Colloid Interf. Sci., 263, p.196 (2003). http://dx.doi.org/10.1016/S0021-9797(03)00028-6
42. Ma, L.; Yan, J.; Gan, M.; Qiu, W.; He, L. & Li, J. - Polym. Test., 27, p.683 (2008). http://dx.doi.org/10.1016/j.polymertesting.2008.05.006
43. Rodrigues, P. C.; Lisboa-Filho, P. N.; Mangrich, A. S. & Aczelrud, L. - Polymer, 46, p.2285 (2005). http://dx.doi.org/10.1016/j.polymer.2005.01.020
44. Ma, J.; Li, Q.; Song, B.; Liu, D.; Zheng, B.; Zhang, Z. & Feng, Y. - J. Mol. Catal. B: Enzym., 56, p.151 (2009). http://dx.doi.org/10.1016/j.molcatb.2008.03.012
45. Stewart, J. P. - J. Mol. Model., 10, p.155 (2004). PMid:14997367. http://dx.doi.org/10.1007/s00894-004-0183-z
46. Salis, A.; Setzu, S.; Mondazzi, M. & Mula, G. - “Porous silicon-based electrochemical biosensors”, in: Biosensors: emerging materials and applications, cap 7, Pier A. Serra (ed.), InTech, Croatia, p.328 (2011).
47. Tuzun, N. S.; Bayata, F. & Sarac, A. S. - J. Mol. Struct.: THEOCHEM, 857, p.95 (2008). http://dx.doi.org/10.1016/j.theochem.2008.02.007
48. Kaur, M. & Datta, M. - “A Novel Methodology for the Total Conversion of Fly Ash into Value Added products: Zeolite. International Conference on Recycling & Reuse of Materials (ICRM 2009)”, Kerala, India (2009).
49. Mishra, A. K. & Tandon, P. - J. Phys. Chem. B, 113, p.9702 (2009). PMid:19569629. http://dx.doi.org/10.1021/jp901143m
50. Reza-Nabid, M.; Golbabae, M.; Bayandori, M.; Dinarvand, R. & Sedghi, R. - Int. J. Electrochem. Sci., 3, p.1117 (2008).
51. Gaudin, F. & Sintes-Zydowicz, N. - Colloid. Surface. A, 331, p.133 (2008). http://dx.doi.org/10.1016/j.colsurfa.2008.07.028
52. Ge, Z.; Zhang, X.; Dai, J.; Li, W. & Luo, Y. - Eur. Polym. J., 45, p.530 (2009). http://dx.doi.org/10.1016/j.eurpolymj.2008.11.008
53. Blinova, N.; Stejskal, J.; Trhová, M.; Prokes, J. & Omastova M. - Eur. Polym. J., 43, p.2331 (2007). http://dx.doi.org/10.1016/j.eurpolymj.2007.03.045
54. Olivucci, M.; Ragazos, N.; Bernardi, F. & Robb, M. A. - J. Am. Chem. Soc., 115, p.3710 (1993). http://dx.doi.org/10.1021/ja00062a042
55. Cantatore, E.; Gelinck, G. H. & De-Leeuw, D. M. - “Polymer electronics: from discrete transistors to integrated circuits and active matrix displays”, in: IEEE Proceedings of the Bipolar/BiCMOS Circuits and Technology Meeting, p.167, Eindhoven, Netherlands (2002).
56. Xu, B.; Choi, J.; Caruso, A. N. & Dowban, P. A. - Appl. Phys. Lett., 80, p.4342 (2002). http://dx.doi.org/10.1063/1.1484247

Received: Aug. 9, 2013
Revised: Apr. 3, 2014
Accepted: Apr. 23, 2014