Zn-Al Mixed Oxides Decorated with Potassium as Catalysts for HT-WGS: Preparation and Properties

Katarzyna Antoniak-Jurak *, Paweł Kowalik, Kamila Michalska, Wieslaw Próchniak and Robert Bicki

Łukasiewicz Research Network—New Chemical Syntheses Institute, Al. Tysiéclecia Państwa Polskiego 13A, 24-110 Puławy, Poland; pawel.kowalik@ins.lukasiewicz.gov.pl (P.K.); kamila.michalska@ins.lukasiewicz.gov.pl (K.M.); wieslaw.prochniak@ins.lukasiewicz.gov.pl (W.P.); robert.bicki@ins.lukasiewicz.gov.pl (R.B.)

* Correspondence: katarzyna.antoniak@ins.lukasiewicz.gov.pl; Tel.: +48-8-1473-1754

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Abstract: A set of ex-ZnAl-LDHs catalysts with a molar ratio of Zn/Al in the range of 0.3–1.0 was prepared using co-precipitation and thermal treatment. The samples were characterized using various methods, including X-ray fluorescence spectroscopy (XRF), X-ray photoelectron spectroscopy (XPS), X-ray powder diffraction (XRD), Fourier transform infrared spectroscopy FT-IR, N2 adsorption, Temperature-programmed desorption of CO2 (TPD-CO2) as well as Scanning electron microscopy (SEM-EDS). Catalyst activity and long-term stability measurements were carried out in a high-temperature water–gas shift (HT-WGS) reaction. Mixed oxide catalysts with various Zn/Al molar ratios decorated with potassium showed high activity in the HT-WGS reaction within the temperature range of 330–400 °C. The highest activity was found for the Zn/Al molar ratio of 0.5 corresponding to spinel stoichiometry. However, the catalyst with a stoichiometric spinel molar ratio of Zn/Al (ZnAl_0.5_K) revealed a higher tendency for surface migration and/or vaporization of potassium during overheating at 450 °C. The correlation of the activity results and TPD-CO2 data show that medium basic sites enhance the progress of the HT-WGS reaction.

Keywords: layered double hydroxides; mixed oxides; spinel structure; potassium promoter; water gas shift

1. Introduction

The water–gas shift (WGS) reaction CO + H2O ↔ CO2 + H2 (ΔH°_298K = −41.1 kJ/mol) is now, and will remain in the future, a very important method for the production of hydrogen [1,2]. The WGS reaction is also of great importance in the production of ammonia throughout steam reforming processes (expressed as CH4 + H2O ↔ CO + 3H2, ΔH°_298K = 205.8 kJ/mol) [3]. Usually, the WGS reaction is performed via a two-stage process to overcome thermodynamic and kinetic limitations [4]. The well-known Fe–Cr–Cu-based catalysts for HT-WGS have numerous significant disadvantages, including the presence of environmentally toxic Cr⁶⁺ with cancerogenic and mutagenic properties. Moreover, in the case of those catalysts, there is also the necessity of conducting the steam reforming process (preceding the HT-WGS process at syngas plants) at a H2O/C ratio not lower than ≈2.8 [5–7]. The decrease in this value, which is very favorable technologically, is connected with the risk of forming iron carbide-catalyzing side reactions (Fischer–Tropsch reactions), as the result of which hydrogen is consumed, causing a number of unfavorable consequences and leading to a decrease in process efficiency. Moreover, these systems are gradually deactivated by thermal recrystallization of Fe₃O₇ enhanced by steam [7]. The development of a new formula of HT-WGS catalyst without the above drawbacks, i.e., without the content of cancerogenic chromium and phyrophoric Fe₃O₇ and having
both high and stable activity and resistance to coking or carbonation at a low steam/gas ratio, is an important area of investigation and long-standing interest for researchers [8,9]. Scientific contributions are mainly focused on the modification of iron-based formulas, i.e., non-stoichiometric magnetite Fe$_{3-x}$O$_{4-y}$ doped with cerium, zirconium, aluminum, copper, and silica oxides [10–13].

Oxide systems (Zn–Al) of specific structures and properties can catalyze the WGS reaction and have many advantages compared to the conventional Fe–Cr–Cu-based systems applied in the discussed reaction. Reviews of the literature indicate that catalysts prepared via hydrotalcite-like (layered double hydroxides—LDHs) precursor routes create particularly interesting opportunities [14–18]. The transition from layered double hydroxides (LDHs) to mixed metal oxides (MMOs) containing spinel oxides is a powerful approach to create materials with tailored properties suitable for various applications. Decomposition of ZnAl-LDHs (>450 °C) leads to the formation of nanocomposite systems of complex structures involving a metal oxide phase (M$_{II}$O), spinel-like phase (M$_{II}$M$_{III2}$O$_{4}$), or systems containing both a spinel structure and single ZnO or Al$_2$O$_3$ (depending on the Zn/Al molar ratio) with unique properties, i.e., highly developed specific surface area, homogeneous distribution of components (even on an atomic level) [19]. These features can make them attractive for numerous catalytic and photocatalytic applications [20,21]. Except from studies on the promoted nickel catalysts of the spinel structure NiFe$_2$O$_4$ [22,23] and other oxides of a spinel structure [24], literature dedicated to alternative systems based on metals other than Fe as an active agent is relatively sparse. The number of reports about applications of HT-WGS catalysts active >300 °C based on simple or mixed oxides (ZnO/ZnAl$_2$O$_4$) is relatively low. To the best of our knowledge, the number of reports concerning the application of (ZnO/ZnAl$_2$O$_4$) catalysts is limited to a few references concerning the reverse water–gas shift [25,26]. Results of our studies conducted so far indicated that catalysts based on mixed oxides (ZnO/ZnAl$_2$O$_4$) promoted with alkali metals (Na, K, Cs) may be considered as attractive alternative of conventional Fe–Cu–Cr applied in industrial practice. The catalytic properties of K-promoted ZnO/ZnAl$_2$O$_4$ and, especially, the effect of the Zn/Al molar ratio on the efficiency and durability of WGS catalysts have not been studied so far.

The objective of this study is to explore the effect of disordered structure of ZnAl$_2$O$_4$ phase in Zn/Al mixed oxide catalysts promoted with potassium on HT-WGS reaction. In particular, the effect of the Zn/Al molar ratio on catalyst activity and thermal stability was investigated. Moreover, the effect of the Zn/Al molar ratio on the structural, textural, and morphological properties of catalysts was investigated. The catalysts were characterized by a variety of techniques, including XRF, XPS, XRD, FT-IR, N$_2$ adsorption, TPD-CO$_2$, and SEM-EDS. Catalytic activity and long-term stability in the HT-WGS reaction were determined in the kinetic regime at conditions similar to the industrial process.

2. Results and Discussion

The results of chemical analysis of the ZnAl$_X$K (fresh) and ZnAl$_X$K$_{AT}$ (after long-term stability tests) catalysts are shown in Table 1. Additionally, in Table 1, the surface concentrations of potassium in the ZnAl$_X$K catalysts, determined by XPS, have been presented. The atomic concentrations of potassium (K$_s$) were converted into weight percentages to compare them with the values obtained from XRF and SEM-EDS analysis (Table 1). The XPS analysis of the fresh ZnAl$_{1.0}$K$_X$ catalyst confirmed a good agreement between the bulk and surface concentrations of potassium. As one can see from Table 1, the values of bulk and surface concentrations of potassium for ZnAl$_{0.3}$K$_X$ and ZnAl$_{0.5}$K$_X$ are not the same. In the case of the ZnAl$_{0.5}$K$_X$ catalyst, the surface concentration of K was higher than the bulk.

The XRF analysis of the ZnAl$_X$K catalysts confirmed a good agreement between the real and nominal Zn/Al molar ratios. In addition, Table 1 shows the composition of potassium (K$_a$) determined by SEM-EDS for the fresh catalysts (ZnAl$_X$K) and for the catalysts after long-term stability tests (ZnAl$_X$K$_{AT}$). For both series, a good agreement between the bulk (K$_b$—determined by XRF) and the content of potassium determined by SEM-EDS (K$_a$) was confirmed.
Table 1. The chemical compositions and textural properties of the ZnAl_X_K and ZnAl_X_K_AT catalysts with different Zn/Al molar ratios.

| Sample (Zn/Al) mol | [wt. %] K | [wt. %] K | S_BET [m²/g] | Vc [cm³/g] | Vm [cm³/g] |
|-------------------|-----------|-----------|--------------|----------|-----------|
| ZnAl_0.3_K        | 0.29      | 3.1       | 2.4          | 154      | 0.53      | 0.48 |
| ZnAl_0.3_K_AT     | 0.27      | 2.8       | 2.7          | -        | 126       | 0.38 | 0.32 |
| ZnAl_0.5_K        | 0.52      | 2.6       | 2.6          | 3.5      | 100       | 0.43 | 0.31 |
| ZnAl_0.5_K_AT     | 0.46      | 2.1       | 2.1          | -        | 77        | 0.34 | 0.25 |
| ZnAl_1.0_K        | 0.95      | 2.9       | 2.8          | 2.7      | 77        | 0.37 | 0.18 |
| ZnAl_1.0_K_AT     | 0.92      | 2.6       | 2.6          | -        | 41        | 0.13 | 0.08 |

* a determined by SEM-EDS; b determined by XRF; c determined by XPS; d total internal surface area determined by Brunauer–Emmett–Teller; e total pore volume determined by the Barret–Joyner–Halend method; f mesopores volume determined by the Barret–Joyner–Halend method.

For the catalysts with a non-stoichiometric Zn/Al molar ratio (ZnAl_0.3_K_AT and ZnAl_1.0_K_AT), after thermal stability tests, slight differences in potassium content were observed. However, they were more significant for the sample of ZnAl_0.5_K_AT. It seems that this can be associated with the higher tendency for surface migration and/or vaporization of potassium during overheating at 450 °C. Comparing the total content of potassium in the ZnAl_X_K_AT series catalysts, determined by the XRF technique, with composition determined by SEM-EDS, it can be concluded that during long-term stability tests, potassium was just slightly evaporated. Similar conclusions come from the work of Bieniasz et al. [27]. According to the authors of this work, alkali promoters may agglomerate or be removed from the catalyst surface due to their volatility at higher temperatures. Moreover, the onset of potassium desorption was observed at 400 °C, meaning that during the HT-WGS process at temperatures below 400 °C, potassium does not escape from the surface of catalyst. When the HT-WGS process was carried out above 400 °C, potassium became mobile.

The comparison of diffraction patterns for ZnAl_X_LDHs precursors with different molar ratio of Zn/Al is presented in Figure 1A, B, respectively. As shown in Figure 1A, the presence of characteristics for the LDH-phase peaks is observed. In the set of diffraction patterns, the first band (in the range 2θ: 11–12°) is the strongest reflex coming from the hydrotalcite-like LDH of the formula Zn_{0.63}Al_{0.37}(OH)_2(CO_3)_{0.185}·xH_2O. However, the intensity of the characteristics bands at angle 2θ: 11.79° (003), 23.69° (006), 34.78° (012), 39.47° (015), 44.27° (018), 55.85° (110) 61.88° (113) decreased with the decreased in Zn/Al molar ratio. In the case of the ZnAl_0.3 material, XRD patterns indicated the presence of a poorly crystallized ZnAl-LDHs phase (Figure 1B). Moreover, it was shown that with the increase in the Zn/Al molar ratio in precursors, the parameter “c” decreased and, at the same time, the parameter “LDH interlamellar distance” increased (see Table S1).

The diffraction patterns for the ZnAl_X_K catalysts are presented in Figure 2. In the diffractograms of ZnAl_0.3_K and ZnAl_0.5_K, the characteristic bands for the cubic spinel structure ZnAl_2O_4 with relatively high crystallinity were evidenced at 2θ: 31.26° (220), 36.68° (311), 44.87° (400), (311), 55.85° (422), 59.35° (511), and 65.35° (440) [28,29]. Due to the structural similarity and similar diffraction images, it is justified to assign reflections for 2θ: 31.80°, 36.10°, and 56.56° corresponding to diffraction on planes (100), (101), and (110), respectively, to the crystalline phase of wurtzite (ZnO) [30].
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Figure 1. (A) XRD patterns of ZnAl-LDHs with different Zn/Al molar ratios; Zn_{0.63}Al_{0.37}(OH)_{2} (CO\textsubscript{3})_{0.185} \cdot xH\textsubscript{2}O phase bands at 2\(\theta\): 11.79° (003), 23.69° (006), 34.78° (012), 39.47° (015), 44.27° (018), 60.50° (110), and 61.88° (113); ZnO phase peaks at angles 2\(\theta\): 31.28° (220), 36.93° (311), 44.62° (400), (100) 36.1° (101). (B) XRD patterns of ZnAl-LDHs with different Zn/Al molar ratios at 2\(\theta\) from 30 to 55°.

For the ZnAl\_1.0\_K sample, spinel phase ZnAl\textsubscript{2}O\textsubscript{4} at 2\(\theta\): 31.28° (220), 36.93° (311), 44.62° (400), 55.30° (422), 59.14° (511), and 65.01° (440) was detected. It is worth noticing that for higher molar ratios of Zn/Al (ZnAl\_1.0\_K), the appearance of an additional phase for angles 2\(\theta\): 34.63°, 47.60°, 56.57°,
62.95° and 68.05°, corresponding to diffraction on planes (002), (102), (110), (103), and (112), respectively, characteristic of a ZnO phase, is visible. It appears that residual zinc oxide which was not incorporated into the spinel structure is present as an individual phase. No phases originating from potassium compounds for the ZnAl_0.3_K and ZnAl_0.5_K samples were evidenced on the XRD patterns. In the case of the ZnAl_1.0_K sample, potassium was evidenced as KNO₃ which was demonstrated by the appearance of weak reflections for angles 2θ: 22.90° and 29.55°. These observations were confirmed by further FT-IR measurements.

The FT-IR spectra of ZnAl-LDHs with variable Zn/Al molar ratios are presented in Figure 3. In all of the samples, three major regions of bands were identified for wavenumbers: 400–800 cm⁻¹, 1550–1250 cm⁻¹, and 3750–2500 cm⁻¹. In the FT-IR spectra, for wavenumbers 3750–2500 cm⁻¹, wide bands with a similar intensity, which are characteristic of stretching bands of O–H in water particles present in interlayer spaces of LDHs and in brucite-like layers, were identified. Moreover, with the increase in the molar ratio of Zn/Al, a shift of bands towards higher wavenumbers (ZnAl_1.0, maximum 3354 cm⁻¹) was observed.

For wavenumbers 1550–1300 cm⁻¹, bands characteristic of antisymmetric stretching vibrations of CO₃²⁻ present for interlayer spaces of LDHs were visible. The intensity of these bands changed with the increase in the Zn/Al molar ratio, with the band of the highest intensity (maximum ~1350 cm⁻¹) at ZnAl_1.0. The contribution of CO₂ in LHDs is in agreement with the data presented in Table S1. For wavenumbers 400–800 cm⁻¹, band characteristics of stretching modes of metal–oxygen–metal bonds, i.e., (Al–O–Al) and (Zn–O–Zn), were present. The most characteristic band differentiating ZnAl_LDHs samples was the band at wavenumbers 700–800 cm⁻¹, present only for ZnAl_1.0_K (maximum 777 cm⁻¹).

This indicates the presence of bending vibrations of O–H groups of H₂O as well as Al–O–Al and Zn–O–H chemical bonds.

In Figure 4, the FT-IR spectra of ZnAl_LDHs_X samples with different Zn/Al molar ratio.

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bands characteristic of stretching vibrations of CO$_3^{2-}$ intercalated between the layers of the rehydrated ex-hydrotalcite-like mixed Zn–Al oxides were evidenced. The CO$_3^{2-}$ stretching vibration became weaker with a slight shift to a higher frequency on decreasing the Zn/Al molar ratio, although it was still indicative of the CO$_3^{2-}$ groups’ decomposition and subsequent CO$_2$ removal from the interlayer space. Additionally, in the ZnAl$_{1.0}$K sample, at 2922 and 2900 cm$^{-1}$, bands characteristic of stretching vibrations of CO$_2$ were detected. In addition, the FT-IR results showed bands of high intensity at 1374–1378 cm$^{-1}$ owing to stretching vibrations of CO$_3^{2-}$ intercalated between the layers of reconstructed ex-hydrotalcite-like mixed Zn–Al oxides.

![FT-IR spectra](image)

**Figure 4.** (A) FT-IR spectra of ZnAl$_X$K catalysts with different Zn/Al molar ratios. (a) ZnAl$_{0.3}$K, (b) ZnAl$_{0.5}$K, (c) ZnAl$_{1.0}$K. (B) The presence of bands characteristic of tetrahedral aluminum (AlO$_4$) and octahedral aluminum (AlO$_6$).

Moreover, for the ZnAl$_{1.0}$K sample, a single peak centered at 1066 cm$^{-1}$ is attributed to NO$_3^-$ anions. For wavenumbers 700–400 cm$^{-1}$, bands characteristic of stretching and bending vibrations of the Al–O bond with octahedral coordination (AlO$_6$) were visible. Bands at 688 cm$^{-1}$ and 533 cm$^{-1}$ corresponded to stretching vibrations of Al–O, whereas the band at 479 cm$^{-1}$ could be attributed to bending O–Al–O bonds with octahedral coordination (AlO$_6$). However, with the increase in the Zn/Al molar ratio, a decrease in the relative intensity of these bands was observed. Moreover, for ZnAl$_{0.3}$K and ZnAl$_{0.5}$K, bands with a low intensity, in the range of wavenumbers 700–850 cm$^{-1}$, coming from Al–O bonds with tetrahedral coordination (AlO$_4$), were identified. The presence of these bonds indicates that the spinel formed under the applied conditions is a zinc-deficient spinel. According to the literature, preparation via coprecipitation and moderate calcinations (<600 °C) of mixed oxides (also with a stoichiometric spinel molar ratio of Zn/Al) usually leads to zinc-deficient spinel [19–21]. In the ZnAl$_{1.0}$K sample, the intensity of the bands attributed to tetrahedral aluminum (AlO$_4$) was higher. The presence of bands characteristic of tetrahedral aluminum (AlO$_4$) indicates the partial inversion of the ZnAl$_2$O$_4$ spinel structure.

Figure 5 shows N$_2$ adsorption–desorption isotherms of the ZnAl$_X$K and ZnAl$_X$KAT samples with different Zn/Al molar ratios. All of the isotherms correspond to H3-type hysteresis according to IUPAC [31], which is a characteristic feature of mesoporous solids. The specific surface area and porous structure parameters of the ZnAl$_X$K and ZnAl$_X$KAT catalysts are given in Table 1. These results show that specific surface areas of the fresh ZnAl$_X$K catalysts were in the range of 77–154 m$^2$/g. With increased Zn/Al molar ratios, significantly lower specific surface areas with a concomitant lower pore volume were observed. A radically lower value of specific surface area (at about 50%) was determined for ZnAl$_{1.0}$K compared to the ZnAl$_{0.3}$K sample. For the ZnAl$_X$KAT series,
a significant decrease in the specific surface area was observed (by 20% as compared to the fresh samples) due to the progressing, sintering, and concomitant pore collapsing. Recrystallization processes were clearly visible for ZnAl\(_{1.0}\)_K, the evidence of which is a decrease in \(S_{\text{BET}}\) by 40% and in total pore volume \(V_c\) by over 60% as compared to fresh sample. The materials of the ZnAl\(_X\)_K\_AT series are characteristic of a lower total pore volume as compared to ZnAl\(_X\)_K ones.

Figure 5. \(\text{N}_2\) adsorption–desorption isotherm of ZnAl\(_X\)_K and ZnAl\(_X\)_K\_AT catalysts with different Zn/Al molar ratios.

Figure S1 shows the pore size distribution of the ZnAl\(_X\)_K and ZnAl\(_X\)_K\_AT samples with different Zn/Al molar ratios.

Pore size distributions prove that pores with a diameter of 20–60 nm account for the majority of the porous structure of the obtained materials. The porous structure comprises three types of pores: mesopores with diameter ranging from 2 to 10 nm, wide mesopores with a dominant diameter of 20–50 nm, and macropores with a dominant diameter of 140–150 nm. The materials of the ZnAl\(_X\)_K\_AT series are characteristic of lower total pore volume as compared to materials of the ZnAl\(_X\)_K series and the dominant pore diameter increased due to overheating.

The surface morphology of the ZnAl\(_X\)_K series (fresh catalysts with different Zn/Al molar ratios) were examined by SEM (Figure 6). Figure 6A,B show that the ZnAl\(_{0.3}\)_K and ZnAl\(_{0.5}\)_K catalysts are built with tightly connected rough quasi-spherical particles and, thus, the hexagonal plate-like morphology typical for LDH materials was not detected. It evidences that the lamellar structure of LDHs was destroyed during thermal treatment. The morphology of the fresh ZnAl\(_{1.0}\)_K catalyst was significantly different (Figure 6C,D). The ZnAl\(_{1.0}\)_K catalyst exhibited a hexagonal platelet morphology of randomly oriented grains. Moreover, aggregates composed of round-edged nanoparticles on their surfaces were also present. It can be seen that, for higher Zn/Al molar ratios, the platelet-like morphology was not totally transformed into a spherical one. However, during the thermal stability test of the ZnAl\(_{1.0}\)_K\_AT catalyst, the hexagonal plates became aggregated and the presence of larger particles with rounded edges is evidenced on SEM images (Figure 6E).
Table S2 shows the peak temperature was observed. Weak basic centers correspond to O–H groups on the surface of the mixed oxides surface which demonstrates the presence of alkali centers with various strengths:

(a) weak centers (α peak); CO$_2$ desorption at 75–180 °C;
(b) medium centers (β peak); CO$_2$ desorption at 180–300 °C;
(c) strong centers (γ peak); CO$_2$ desorption above 300 °C.

Moreover, for the ZnAl$_{0.5}$K catalyst, the shift of bands (α, β, γ) towards higher maximum peak temperature was observed. Weak basic centers correspond to O–H groups on the surface of the catalysts; medium strength sites are associated to the oxygen in Zn$^{2+}$–O$^{2-}$, Al$^{3+}$–O$^{2-}$ and K$^+$–O$^{2-}$ pairs, and strong basic sites correspond to low coordination oxygen atoms [32]. Table S2 shows the contribution of basic sites to ZnAl$_X$K samples and $T_{\text{max}}$ desorption. Deconvolution results (see Table S1 and Figure 7) show that medium centers (β peak) are a dominant in all the tested samples.

For ZnAl$_{0.3}$K and ZnAl$_{1.0}$K, similar levels of CO$_2$ desorption are observed which correspond with weak alkali centers (α) and medium centers (β). Concomitantly, for ZnAl$_{1.0}$K, the increase in concentration of strong alkali centers (γ) on the catalyst surface was observed as compared to ZnAl$_{0.3}$K. For the sample with a stoichiometric spinel molar ratio of ZnAl (ZnAl$_{0.5}$K), the desorption of a larger (compared to the other two samples) amount of CO$_2$ at 180–400 °C was observed, which indicates that the number of surface active sorption sites is higher for stoichiometric ZnAl$_2$O$_4$ spinel compared to other studied systems. Moreover, it seems reasonable that the excessive amount of free ZnO hinders CO$_2$ adsorption on medium alkali centers present on the surface of the ZnAl$_{1.0}$K catalyst. It also seems reasonable to assume that CO$_2$ is less efficiently adsorbed on the zinc-deficient spinel ZnAl$_{0.3}$K for the entire range of studied temperatures.

Figure 8A shows kinetic data for the set of ZnAl$_X$K fresh catalysts with different Zn/Al molar ratios. The results reveal that the tested catalysts exhibited high catalytic activity in the HT-WGS reaction.
reaction at temperatures up to 400 °C. The activity of the ZnAl_0.3_K and ZnAl_1.0_K catalysts containing a defective spinel phase decorated with potassium was high. However, the highest activity was revealed for the sample with a stoichiometric spinel molar ratio of Zn/Al (ZnAl_0.5_K). Moreover, the ZnAl_0.5_K catalyst exhibited relatively high activity at a low temperature range of 330–350 °C. The catalytic activity of the ZnAl_X_K catalysts in the HT-WGS reaction is associated with the surface basicity. This confirms well-evidenced observations indicating a correlation between the number of medium basic sites and catalytic activity in the HT-WGS reaction (see TPD-CO2 results).

The highest activity of ZnAl_0.5_K in the HT-WGS reaction is caused by the presence of larger (compared to the other two samples) amount of medium basic centers (see Table S2). Moreover, the ZnAl_0.5_K catalyst exhibited relatively high activity at a low temperature range of 330–350 °C. The catalytic activity of the ZnAl_X_K catalysts in the HT-WGS reaction is associated with the surface basicity. This confirms well-evidenced observations indicating a correlation between the number of medium basic sites and catalytic activity in the HT-WGS reaction (see TPD-CO2 results).

Figure 7. TPD_CO2 profiles of ZnAl_X_K samples with different Zn/Al molar ratios.

Figure 8. Arrhenius plots for high-temperature water–gas shift (HT-WGS) on ZnAl_X_K (A) and on ZnAl_X_KAT catalysts (B) at 330, 350, 370, and 400 °C.

The highest activity of ZnAl_0.5_K in the HT-WGS reaction is caused by the presence of larger (compared to the other two samples) amount of medium basic centers (see Table S2). Moreover, the reasons for lower catalyst activity of ZnAl_1.0_K may be associated with partially disordered structure of ZnAl2O4 phase in Zn–Al mixed oxide catalysts decorated with potassium. According to the FT-IR results (see Figure 4A,B), the occurrence of characteristic tetrahedral coordination bands (AlO4) indicates a partial inversion of the ZnAl2O4 spinel structure.

Despite the high activity, the K-decorated Zn–Al systems differ in thermal stability depending on the Zn/Al molar ratio (Figure 8B). The decrease of activity by ca. 20% and 40% for ZnAl_0.5_K and
ZnAl\(_{1.0}\) K, respectively, was observed as a result of the samples being overheated in HT-WGS process conditions. This may be caused by recrystallization (see Table 1) during the catalyst performance. In spite of recrystallization, the ZnAl\(_{1.0}\) K catalyst maintained high catalytic activity in the HT-WGS process. The best thermal stability was found for the catalyst with a stoichiometric spinel phase as the dominant one (ZnAl\(_{0.5}\) K\(_{\text{AT}}\)), despite the visible escape of potassium.

Table 2 shows the HT-WGS reaction rate (r) of the ZnAl\(_{X}\) K and ZnAl\(_{X}\) K\(_{\text{AT}}\) catalysts with different Zn/Al molar ratios. Additionally, the activity of Fe-based reference HT-WGS catalysts has been presented. The results reveal that, for the tested range of temperatures, the ZnAl\(_{0.5}\) K catalyst showed higher activity by approx. 20% as compared to the (Fe–Cr–Cu) reference HT-WGS catalyst. A comparable activity level was observed for ZnAl\(_{1.0}\) K. A lower activity, as compared to Fe–Cr–Cu, was observed for ZnAl\(_{0.3}\) K (activity lowered by approx. 10% at 400 °C).

The comparison of data collected in Table 2 also shows the substantial differences of \(E_a\) between the Fe-based reference catalysts and the series of ZnAl\(_{X}\) K and ZnAl\(_{X}\) K\(_{\text{AT}}\) catalysts. As shown in the presented data, the ZnAl\(_{X}\) K are characteristic of clearly lower \(E_a\) values as compared to the Fe–Cr–Cu (162.2 ± 1.1) applied in the HT-WGS. The presented data for ZnAl\(_{X}\) K in the kinetic regime are 91.8–115.3 kJ/mol. For the ZnAl\(_{X}\) K\(_{\text{AT}}\) catalysts, slightly lower \(E_a\) values were observed at 99.4–118.74 kJ/mol.

| Sample          | 330 °C | 350 °C | r[Nm\(^3\)CO g\(^{-1}\)h\(^{-1}\)] | 370 °C | 400 °C | \(E_a\) [kJ/mol] |
|-----------------|--------|--------|-------------------------------------|--------|--------|------------------|
| ZnAl\(_{0.3}\) K | 0.40   | 0.52   | 1.08                                | 1.87   | 91.8 ± 0.2   |
| ZnAl\(_{0.5}\) K | 0.41   | 0.87   | 1.43                                | 2.65   | 106.5 ± 0.1  |
| ZnAl\(_{1.0}\) K | 0.26   | 0.57   | 0.98                                | 2.15   | 115.3 ± 0.3  |
| ZnAl\(_{0.5}\) K\(_{\text{AT}}\) | 0.14   | 0.39   | 0.56                                | 1.00   | 99.4 ± 0.2   |
| ZnAl\(_{1.0}\) K\(_{\text{AT}}\) | 0.26   | 0.61   | 0.95                                | 2.21   | 118.4 ± 0.1  |
| ZnAl\(_{0.3}\) K\(_{\text{AT}}\) | 0.19   | 0.41   | 0.59                                | 1.40   | 103.6 ± 0.1  |
| Fe–Cr–Cu \(^*\) | 0.26   | 0.67   | 1.33                                | 2.10   | 162.2 ± 1.1  |

\(^*\) Fe–Cr–Cu—commercial HT-WGS catalyst.

3. Materials and Methods

3.1. Materials and Synthesis

A series of Zn–Al–LDH precursors with different molar ratios of Zn/Al were prepared by co-precipitation. Zn–Al–LDH precursors with nominal compositions of (Zn/Al)\(_{\text{mol}}\) = 0.3, 0.5, and 1.0 were prepared by a simultaneous dosing of two streams of Zn(NO\(_3\))\(_2\)·6H\(_2\)O (Chempur, Piekary Śląskie, Poland), Al(NO\(_3\))\(_3\)·9H\(_2\)O (99% purity, Chempur, Piekary Śląskie, Poland) aqueous solution and Na\(_2\)CO\(_3\)/NaOH (99% purity, Chempur, Piekary Śląskie, Poland) aqueous solution to the precipitation reactor under vigorous stirring. Synthesis was carried out at a temperature of 75–80 °C and pH in the range of 7.5–8. After complete precipitation, the slurry was aged at 80 °C for 0.5 h under vigorous stirring. In the next step, the sodium and nitrate ions were thoroughly removed by washing with redistilled water to reach a filtrate conductivity below 150 µS/cm, and were then separated, dried, and calcined at 500 °C for 4 h. The obtained solids were ground to 0.16–0.25 mm fraction. In the next stage, the materials were subjected to incipient wetness impregnation with an aqueous solution of potassium nitrate (99.0% purity, Sigma Aldrich, Poznań, Poland) with concentrations selected on the basis of the water absorption of Zn–Al oxides, allowing to obtain catalysts with a nominal K content equal to 3 wt.%. After impregnation, the samples were dried and recalcined at 450 °C for 2 h. The resulting solids were indicated: ZnAl\(_{X}\) K, where X is the Zn/Al molar ratio.
3.2. Characterization

The chemical composition of the samples was determined by the WD XRF method using a Axios mAx spectrometer (Malvern Panalytical, Malvern, UK), equipped with a lamp Rh SST-mAx 4 kW. The specific surface area and total pore volume were determined, using ASAP® 2050 Xtdented Pressure sorption analyzer (Micromeritics Instrument Co., Norcross, GA, USA), from N₂ adsorption and desorption isotherms at a temperature of -196 °C for p/p₀ ≤ 0.99, using the BET adsorption model and BJH (Barret–Joyner–Halenda) transformation for mesopore characteristics. The XRD measurements were performed on an Empyrean (Malvern Panalytical, Malvern, UK) system (Bragg-Brentano geometry) equipped with a PIXcel3D detector, using Cu Kα radiation (λ = 1.542 Å) and operating at 40 kV and 40 mA. The crystallite size of the Zn–Al–LDH’s phase was calculated using Scherrer’s formula from the diffraction broadening on planes (003), (006), (012), and (015). Fourier-transform infrared spectroscopy (FT-IR) measurements were taken from Nicolet 8700A (Thermo Scientific, USA) equipped with a diamond crystal (Smart Orbit TR diamond ATR). Infrared spectra were obtained in a 4000–400 cm⁻¹ spectral range. A high resolution FEI Quanta3D FEG microscope (Waltham, MA, USA) with an electron beam energy of 20 kV was used in order to analyze the surface morphology by scanning electron microscopy (SEM) using an EDX Octane Elect Plus detector. Analysis of surface composition was carried out for 5 areas (150 µm × 150 µm). The specified surface composition was the average derived from results for five randomly chosen separated sub-areas. The surface basicity of the ZnAl_X_K catalysts was measured by temperature programmed desorption of CO₂ (TPD-CO₂) using a AutoChem 2950 HP analyzer (Micromeritics Instrument Co., Norcross, GA, USA). Each sample (0.2 g) was pre-treated in a He flow at 500 °C for 30 min (10 °C/min); the samples were then cooled to 200 °C and CO₂ adsorption was performed, and then the samples were cooled to 150 °C. In the next stage, the CO₂ stream was replaced with a He purge to remove the physisorbed CO₂ and samples were cooled down to 20 °C. Finally, the desorption process was performed at a heating rate of 10 °C/min to 500 °C. Surface elemental analyses were performed by X-ray photoelectron spectrometry (multi-chamber analytical system UHV, Prevac, Rogów, Poland) at a base pressure better than 1.0 × 10⁻⁸ mbar, using a polychromatic Al K X-ray source (1486.7 eV).

3.3. Methods

The catalysts’ activity and thermal stability in the HT-WGS reaction were carried out in kinetic regime using a 4-channel differential reactor (at a conversion degree below 0.2). HT-WGS measurements were performed under the following conditions: catalyst grain size (mm), 0.16–0.25; pressure (MPa), 2.5; temperature (°C), 330–400; gas flow (Ndm³/h), 60; dry gas composition (% vol.); CO/CO₂/H₂ = 3.0/10.0/87.0; H₂O/gas molar ratio = 1.

The HT-WGS reaction rate at the kinetics regime was calculated based on a mass balance of the reactor:

\[ r = 0.01 \cdot V \cdot \Delta C_{CO} \cdot \frac{m_{cat}}{1} \]  

(1)

where:

r—measured HT-WGS reaction rate in the kinetics regime (Ndm³CO/(g cat·h⁻¹)),
V—dry gas flow [Ndm³·h⁻¹],
ΔC_CO—the difference between inlet and outlet CO content in a dry gas [vol. %],
m_cat.—weight of the catalyst sample.

The HT-WGS reaction rate constant was calculated from the equation:

\[ r = k \cdot p_{CO}^{1.04} \cdot p_{CO_2}^{-0.25} \cdot p_{H_2O}^{0.16} \cdot p_{H_2O} \cdot (1 - \beta) \]  

(2)

where:

r—HT-WGS reaction rate [Ndm³CO/(g_cat·h)],
k—HT-WGS rate constant \([\text{Ndm}^3\text{CO}/(\text{gcat}\cdot\text{h}\cdot\text{at})^{0.95}]\),
pCO, pCO\(_2\), pH\(_2\), pH\(_2\)O—partial pressure of CO, CO\(_2\), H\(_2\), H\(_2\)O [ata],
\(\beta = p\text{CO}_2 \cdot pH_2 \cdot K_p^{-1} \cdot p\text{CO}^{-1} \cdot p\text{H}_2\text{O}^{-1}\)—approach to equilibrium,
K\(_p\)—WGS equilibrium constant.

The catalytic effect’s durability was evaluated on the basis of activity changes as a result of overheating the catalysts at 450 °C over 24 h in a stream of process gas. The ZnAl\(_X\)K catalysts, after long-term stability tests, were indicated as ZnAl\(_X\)K\(_{AT}\), where X is the Zn/Al molar ratio.

4. Conclusions

Mixed oxide catalysts (ZnO/ZnAl\(_2\)O\(_4\)) derived from LDH precursors and K-decorated showed high specific surface areas. Increasing the Zn/Al molar ratio resulted in decreases in the specific surface area and total pore volume of catalysts. The amount and strength of surface basic sites on ex-ZnAl-LDH decorated with potassium depended on the (Zn/Al)\(_{mol}\) ratio. The correlation of activity results and TPD-CO\(_2\) data show that medium basic sites enhance HT-WGS reaction.

The catalytic activity of ZnAl\(_X\)K in the HT-WGS reaction was relatively high and depended on the Zn/Al molar ratio. The highest activity was observed for the catalyst with a stoichiometric spinel molar ratio of Zn/Al (ZnAl\(_0.5\)K). The catalyst containing a zinc-deficient spinel phase decorated with potassium (ZnAl\(_0.3\)K) and the catalyst with partial inversion of the ZnAl\(_2\)O\(_4\) spinel phase in Zn–Al mixed oxide decorated with potassium (ZnAl\(_1.0\)K) showed lower catalytic activity in HT-WGS process conditions.

Sintering of ZnO-ZnAl\(_2\)O\(_4\) mixed oxides, with a concomitant loss of platelet morphology, occurred even during the preparation of calcinations step (500 °C). Nevertheless, in spite of recrystallization as a result of overheating, the ZnAl\(_1.0\)K catalyst maintained high catalytic activity in the HT-WGS process. The catalyst with a stoichiometric spinel molar ratio of Zn/Al (ZnAl\(_0.5\)K showed a higher level of activity by about 20% compared to the Fe–Cr–Cu catalyst.

Supplementary Materials: The following are available online at http://www.mdpi.com/2073-4344/10/9/1094/s1:
Table S1: The comparison of the physicochemical properties and lattice parameters of ZnAl\(_X\)LDHs; Table S2: Contribution of basic sites for ZnAl\(_X\)K catalysts. Peak area ratio is defined as the ratio of the sum of \(\alpha\), \(\beta\) and \(\delta\) peak area of ZnAl\(_X\)K catalysts to ZnAl\(_0.5\)K catalyst; Figure S1: Pore size distribution of ZnAl\(_X\)K and ZnAl\(_X\)K\(_{AT}\) samples with different Zn/Al molar ratios.

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Conflicts of Interest: The authors declare no conflict of interest.

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