Controlling molecular assembly and charge transport of n-type organic semiconductors with sterically demanding substituents

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Abstract

Benzo[de]isoquinolo[1,8-gh]quinolinetetracarboxylic diimide (BQQDI) n-type organic semiconductors demonstrate unique multi-fold intermolecular hydrogen-bonding interactions that lead to excellent aggregated structures, charge transports, and electron mobility. However, the robust intermolecular anchoring of BQQDI presents challenges for further fine-tuning molecular assemblies and organic semiconductor properties. Herein, we report the design and synthesis of two BQQDI derivatives with sterically demanding phenyl- and cyclohexyl-substituted BQQDI (Ph–BQQDI and Cy₆–BQQDI), where the two organic semiconductors show distinct molecular assemblies and degrees of intermolecular orbital overlaps. In addition, the difference in their packing motifs led to strikingly different band structures that give rise to contrasting charge-transport capabilities. As a result, Cy₆–BQQDI shows excellent transistor performances in both single-crystalline and polycrystalline thin-film organic field-effect transistors.

Introduction

Charge transport that gives rise to electrical properties of organic semiconductors (OSCs) is typically governed by intermolecular orbital overlaps and controlling such intermolecular interactions to achieve effective charge-transport properties lies in the center of molecular design for high-performance OSCs¹². In the past decades, intense investigations of high-performance OSCs in terms of molecular design and device engineering fueled the rapid development of applicable organic-based electronic devices such as organic field-effect transistors (OFETs)³–⁵, which offer mechanical flexibility and low-cost processing compared to traditional inorganic-based devices. In particular, the hole-transporting p-type OSCs have shown promising OFET performances with charge-carrier mobilities (µ) over 10 cm² V⁻¹ s⁻¹, and not only do these materials lead to applicable devices, they also provided crucial information on charge transport and guidance for future molecular designs⁶–¹³. On the other hand, the electron-transporting n-type OSCs, which are an essential component for constructing organic-based logic circuits¹⁴–¹⁶, are generally inferior to state-of-the-art p-type OSCs in terms of µ. One of the challenges associated with the molecular design of n-type OSCs is that, owing to the energetics of charge injections and carrier transport, the lowest unoccupied molecular orbital (LUMO) level of n-type OSCs needs to be sufficiently low (< −4.0 eV) to protect charge carriers from oxidation by ambient oxygen and moisture¹⁷,¹⁸. While the air-stability issue of n-type OSCs can be addressed by incorporations of electron-deficient moieties¹⁹–²¹ and several studies have reported air-stable n-type OSCs with encouraging OFET
performances\textsuperscript{22–25}, design strategies that focus on effective intermolecular orbital overlaps (quantified by transfer integral $t$ and effective mass $m^\ast$)\textsuperscript{26,27} and molecular assemblies for achieving favorable charge-transport properties and high electron mobility ($\mu_e$) are still required.

Fig. 1 Molecular features of PhC$_2$–BQQDI. a Structural, packing motif and charge transport comparisons between PDI and BQQDI; b Intermolecular interactions of PhC$_2$–BQQDI; c Molecular misalignment of PhC$_2$–BQQDI.

Recently, our group reported an air-stable and high-performance benzo[de]isoquinolino[1,8-$gh$]quinolinetetracarboxylic diimide (BQQDI) $\pi$-electron core\textsuperscript{28} ($\pi$-core) (Fig. 1a). The BQQDI is structurally analogous to the widely studied perylenetetracarboxylic diimide (PDI) system (Fig. 1a), though the electronegative nitrogen atoms in the BQQDI framework result in a DFT calculated deep-lying LUMO level of $\SI{-4.17}{\text{eV}}$ (at the B3LYP/6-31G+(d) level)\textsuperscript{29} for potential air-stable $n$-type charge transports, whereas the PDI $\pi$-core possesses a shallower LUMO level of $\SI{-3.80}{\text{eV}}$. Upon functionalization of the BQQDI $\pi$-core with phenethyl (PhC$_2$–BQQDI) groups, multi-fold hydrogen-bonding interactions are formed between adjacent molecules in
the transverse direction (Fig. 1b), and strong π-π interactions are also observed in the vertical direction. The resulting brickwork packing motifs show large \( t \) values which indicate two-dimensional (2D)-like charge-transport properties, whereas simple PDI derivatives generally exhibit one-dimensional (1D) π-π stacking motif\(^{30,31}\) that leads to anisotropic charge-transport capabilities. The PhC\(_2\)–BQQDI derivative, forms favorable phenyl-to-phenyl edge-to-face interactions between each molecular layer (Fig. 1b), in addition to the aforementioned intermolecular features, which significantly reinforce the intermolecular orbital overlaps as well as suppression of molecular motions. As a result, PhC\(_2\)–BQQDI exhibits an impressive \( \mu_e \) of 3.0 cm\(^2\) V\(^{-1}\) s\(^{-1}\) in solution-processed OFETs, and excellent robustness against thermal- and bias-stress, which are necessary features for practical organic electronic applications.

Despite the encouraging results of PhC\(_2\)–BQQDI-based OSCs, the robust core-to-core and interlayer intermolecular interactions also pose challenges to further fine-tune molecular assemblies and charge-transport properties of BQQDI derivatives. By examining the packing structure of PhC\(_2\)–BQQDI, we notice that the hydrogen-bonding interactions cause some degree of misalignment in the π-π stacking direction, which lead to an unbalanced charge-transport capability reflected by its \( t \) and \( m^* \) values. Herein, we report the investigation of two BQQDI derivatives with phenyl (Ph–BQQDI) and cyclohexyl (Cy\(_6\)–BQQDI) substituents on their molecular assemblies and charge-transport capabilities. From a chemical perspective, we envisage that the installment of these sterically demanding substituents close to the BQQDI π-core compared to PhC\(_2\)–BQQDI may sufficiently weaken the hydrogen-bonding interactions in the transverse direction and reduce the misalignment in intermolecular orbital overlaps. Owing to the different geometric and electronic properties, Ph–BQQDI and Cy\(_6\)–BQQDI exhibit distinct intra and interlayer molecular assemblies that lead to contrasting charge-transport capabilities as well as OSC performances.
Results and Discussion

**Fig. 2 Synthetic routes for BQQDI derivatives.** a Synthesis of Ph–BQQDI from BQQ–TCDA; b Synthesis of Cy₆–BQQDI from BQQ–TCP.

The first target compound Ph–BQQDI was synthesized from the benzo[de]isoquinolino[1,8-gh]quinolinetetracarboxylic dianhydride (BQQ–TCDA) starting material according to the previously reported procedure²⁸ in 91% yield (Fig. 2a). However, formation of Cy₆–BQQDI could only reach 70% from BQQ–TCDA, along with 5% mono-functionalized intermediate and 25% remaining starting material, likely due to lowering of the reactivity of amine by the sterically demanding cyclohexyl moiety. As reported before, the synthesis of 4-heptyl-substituted BQQDI (4-Hep–BQQDI) with bulky branched alkyl chains afforded only 10% yield by using BQQ–TCDA as the starting material. To circumvent this issue, we discover that the precursor of BQQ–TCDA, 3,9-dimethyl-4,10-bis(2,4,6-trichlorophenyl)benzo[de]isoquinolino[1,8-gh]quinoline-3,4,9,10-tetracarboxylate (BQQ–TCP) can also act as a viable starting material for the synthesis of BQQDI derivatives. The electrophilic trichlorophenyl formate groups³² of BQQ–TCP provides high reactivity towards imide condensation and tolerance to the somewhat bulky cyclohexyl amine. The Cy₆–BQQDI target compound was successfully furnished from BQQ–TCP in 94% yield (Fig. 2b), and 4-Hep–BQQDI was also generated in 86% yield using the same procedure.
Fig. 3 Molecular assemblies and charge-transport capabilities of Ph–BQQDI and Cy6–BQQDI. a Dimer distances, interactions and LUMO overlaps; b Interlayer interactions; c Stacking distances and transfer integrals.

The single crystals of Ph–BQQDI and Cy6–BQQDI were prepared using physical vapor transport (PVT) and solution-grown methods, respectively, and large plate-like crystals were obtained for both compounds (Fig. S5). Ph–BQQDI crystallizes in the monoclinic $P2_1/c$ space group with a 2D brickwork packing motif. Each planar BQQ $\pi$-core forms a four-fold hydrogen-bonding interactions with O···H and N···H short contacts on each side with its adjacent molecules in the transverse direction, along with slipped $\pi$-$\pi$ stacking interactions (Fig. 3a). The substituents of Ph–BQQDI form multiple phenyl-to-phenyl interactions in the interlayer space (Fig. 3b), which has been shown to be a favorable feature for suppressing molecular fluctuations$^{28}$. Within the brickwork assembly of Ph–BQQDI, distances of the $\pi$-
stacks are found to be 3.36 Å and 3.37 Å (Fig. 3c), and the slight difference in distances is attributed to the misalignment between adjacent molecules in the transverse direction. The molecular assembly of Ph–BQQDI leads to a misalignment of LUMO of molecules in the π-π stacking direction (Fig. 3a), where only a small degree of LUMO overlaps is observed between the top molecule and the molecule in the bottom layer. By calculating the t values of Ph–BQQDI based on its crystal structure, it is evident that the misalignment in the assembly leads to different degrees of orbital overlaps with \( t_1 \) and \( t_3 \) equal to +77.23 meV and +48.57 meV, respectively (Fig. 3c). Strong transverse interactions between π-cores are quantified by \( t_2 \) values of +18.55 meV. By comparing the \( t \) values of Ph–BQQDI with the high-performance PhC\(_2\)–BQQDI (\( t_1 = +90.71 \) meV, \( t_2 = +18.89 \) meV, \( t_3 = +58.47 \) meV), the \( t \) values of Ph–BQQDI are much smaller than those of PhC\(_2\)–BQQDI, especially in the π-π stacking direction.

The Cy\(_6\)–BQQDI derivative crystallizes in the monoclinic C2/m space group with more symmetry than that of Ph–BQQDI. The steric bulk of cyclohexyl substituents likely prevents close contact between molecules in the transverse direction, which shows O···H and N···H interactions of 2.93 Å and 3.05 Å, respectively, that are larger in distances than those between Ph–BQQDI dimers. Force constant calculations of the transverse dimers at the M06-2X/6-31++G(d,p) level further substantiate that Cy\(_6\)–BQQDI shows a much weaker interaction energy of \(-2.20 \) kcal mol\(^{-1}\) than that of Ph–BQQDI (\(-7.15 \) kcal mol\(^{-1}\)) (Fig. 3a). However, transverse dimers of Cy\(_6\)–BQQDI show a much smaller displacement in the long molecular axis direction than Ph–BQQDI dimers, and the reduced molecular misalignment of Cy\(_6\)–BQQDI leads to a much more enhanced LUMO overlaps in the π-π stacking directions. In contrast to Ph–BQQDI, Cy\(_6\)–BQQDI does not show any apparent interactions in the interlayer space (Fig. 3b), and we speculate that this may lead to larger molecular fluctuations of Cy\(_6\)–BQQDI than Ph–BQQDI. The 2D brickwork motif of Cy\(_6\)–BQQDI shows a uniform π-π stacking distance of 3.33 Å, which corresponds to the same degree of intermolecular orbital overlap with \( t_1 = t_3 = +88.28 \) meV, which are larger than those of Ph–BQQDI. Even though the transverse dimer of Cy\(_6\)–BQQDI demonstrates much weaker interaction energy than that of Ph–BQQDI dimer, the transverse intermolecular orbital overlap of Cy\(_6\)–BQQDI that is quantified by \( t_2 \) (+14.27 meV) is only slightly lower than that of Ph–BQQDI (+18.55 meV) (Fig. 3c). The uniform charge-transport capability exhibited by Cy\(_6\)–BQQDI may indicate promising OSC performances. 

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Fig. 4 Molecular simulations of Ph–BQQDI and Cy₆–BQQDI. Color-coded B-factor distributions obtained from the trajectories during the last 10 ns of a 100 ns MD simulations in the NTP ensemble.

The different interlayer interactions of Ph–BQQDI and Cy₆–BQQDI prompted us to investigate their molecular fluctuations in the single-crystal state, and MD simulations with constant amount of substance (N), temperature (T), and pressure (P) (isothermal-isobaric NTP ensemble) are performed based on their single-crystal structures measured at room temperature. Ph–BQQDI with multiple phenyl-to-phenyl interactions in the interlayer space shows B-factor distributions similar to PhC₂–BQQDI, where small molecular fluctuations are observed on the substituents as well as the π-cores (Fig. 4). However, the atoms on the ortho and bay positions of Ph–BQQDI molecules show slightly larger B-factors than the rest of the π-core, which indicate that the rigid phenyl substituents and multiple interlayer interactions may not necessarily contribute to suppressed molecular fluctuations, rather the lack of flexibility in the substituents may contribute to some degree of structural perturbation. The B-factors of Cy₆–BQQDI are also similar to those of PhC₂–BQQDI and Ph–BQQDI, where large B-factors are not observed on the ortho and bay position atoms (Fig. 4). Though, Cy₆–BQQDI shows a unique case where some molecules have larger B-factors than others, and such a variant B-factor distribution might be contributed to its lack of interlayer interactions as well as weak intermolecular interactions in the transverse direction.
Fig. 5 OFET performances and thin film assemblies of Ph–BQQDI (left) and Cy₆–BQQDI (right). a Transfer characteristic of single-crystalline transistors; b Molecular assemblies in device states and corresponding channel directions.

To evaluate the $\mu_e$ of Ph– and Cy₆–BQQDI, we fabricated bottom-gate/top-contact OFETs with gold electrodes using their single-crystalline thin films as the active OSC layer. Owing to the poor solubility of Ph–BQQDI, the PVT-grown single-crystalline thin films (193 nm thick) were directly laminated on a silicon substrate coated with a parylene insulating polymer, which has been used for laminated single-crystal OFETs. The OSC single-crystalline thin films (7.79 nm thick) of the more soluble Cy₆–BQQDI were prepared by the edge-casting method on the AL-X601-coated silicon substrate, which is an excellent insulating layer for solution-processed n-type single-crystalline thin-film OFETs. The maximum $\mu_e$ of Ph–BQQDI was measured to be 1.0 cm² V⁻¹ s⁻¹ (Fig. 5a). The highest $\mu_e$ of 2.3 cm² V⁻¹ s⁻¹ was achieved by the single-crystalline thin-film OFET using Cy₆–BQQDI and an average $\mu_e$ of 1.8 ± 0.21 cm² V⁻¹ s⁻¹ was measured over 12 devices (Fig. S7), and the devices showed excellent air-stability over one month (Fig. S8). Thin-film X-ray diffractions of the OSC active layers of Ph– and Cy₆–BQQDI reveal that their OFET channels directions correspond to the $b$-crystallographic axis and the [1 1 0] direction, respectively (Fig. S6). The molecular stacks of Cy₆–BQQDI are
roughly orthogonal to the OFET substrate with the π-π stackings that are parallel to the electron transport, whereas the molecular assembly of Ph–BQQDI creates more of an offset between the electron transport and the π-π stacking direction, which possibly leads to a less efficient electron transport.

Polycrystalline thin-film (40 nm thick) OFETs of Ph– and Cy₆–BQQDI were also fabricated via vacuum deposition using decyltrimethoxysilane (DTS) as the self-assembled monolayer (SAM). The deposited thin film of Ph–BQQDI does not assume its single-crystal structure, as the d-spacing of 19.5 Å at 2θ = 4.52˚ (d-spacing = 15.5˚ in single crystal) indicates a tilting angle of 24.5˚ between the long axis of the molecules and the substrate (Fig. S11 and S13), which possibly originates from the interactions between the substrate and OSC molecules. The polycrystalline thin film of Cy₆–BQQDI on the other hand, shows consistent molecular assembly as its single-crystal structure. Though, the diffraction peak at 2θ = 17.04˚ corresponds to the (11-1) plane of the single-crystal structure of Cy₆–BQQDI, which indicates a thin-film orientational disordering with both edge-on and face-on-like stackings (Fig. S11 and S14).

We evaluated the polycrystalline thin-film OFETs of Ph– and Cy₆–BQQDI, and the highest μₑ of 0.16 cm² V⁻¹ s⁻¹ was obtained for Ph–BQQDI (Fig. S15), which is one-order lower than its single-crystalline device, likely due to large grain boundary of the polycrystalline thin films (Fig. S10). Cy₆–BQQDI-based polycrystalline OFETs afforded the highest μₑ of 0.50 cm² V⁻¹ s⁻¹ on DTS (Fig. S16), and this promising result motivated us to explore other device conditions.

When changed the SAM from DTS to hexamethyldisilazane (HMDS), which decreases the ratio of face-on/edge-on assemblies (Fig. S12), the highest μₑ of 40 nm-thick polycrystalline devices of Cy₆–BQQDI was further improved to 0.66 cm² V⁻¹ s⁻¹ (Fig. S14). We found that by reducing the Cy₆–BQQDI OSC layer thickness from 40 nm to 20 nm on HMDS, an excellent highest μₑ of 1.0 cm² V⁻¹ s⁻¹ could be achieved, and the μₑ appeared to be independent of the channel length (100-500 µm) (Fig. S17-19). The device performances of polycrystalline Cy₆–BQQDI on DTS and HMDS in air are also consistent over more than one month (Fig. S20).

The μₑ of polycrystalline Cy₆–BQQDI is one of the highest among current BQQDI derivatives, though, we speculate that its overall polycrystalline device performance might be hampered by the orientational disordering of its thin-film molecular assembly, and further optimization of the deposition conditions is currently undergoing. Both single- and polycrystalline OFETs based on Cy₆–BQQDI show significantly higher μₑ than those based on Ph–BQQDI, and the difference in their device performances are in agreement with their calculated t values, but more in-depth analysis of their charge-transport capabilities is required.
Fig. 6 Charge-transport capabilities calculated by the tight-binding approximation. a and d 2D HOMO band; b and e Angle-resolved effective mass; c and f Angle-resolved inversed effective mass of Ph–BQQDI and Cy₆–BQQDI, respectively.

We then further investigated the directionality of the charge-transport capabilities of Ph– and Cy₆–BQQDI using the tight-binding approximation to rationalize the difference in their OFET performances. Ph–BQQDI exhibits elliptical shaped 2D LUMO bands, and from the bottom edge of the band dispersion (Fig. S21), we calculated the \( m^* \) values with respect to the crystallographic axes. The smallest \( m^* \) value of Ph–BQQDI is 1.6 \( m_0 \), which is found at 45° from the \( a \)-axis ([1 1 0] direction). However, the OFET channel along the \( b \)-axis direction of Ph–BQQDI corresponds to a larger \( m^* \) of 2.4 \( m_0 \). On the other hand, Cy₆–BQQDI shows a more circular 2D LUMO band and the resulting \( m^* \) values are seemingly uniform across the crystallographic directions. The \( m^* \) of Cy₆–BQQDI along the OFET channel direction is calculated to be 1.9 \( m_0 \), which is only slightly larger than its smallest \( m^* \) of 1.8 \( m_0 \) in the \( a \)-axis direction. The directionality of the charge transport of the current BQQDI derivatives can be better visualized from the angle-resolved inversed \( m^* \) plots, where Ph–BQQDI shows a peanut shaped curve, with its best charge transport at 45° relative to the \( a \)-axis ([1 1 0] direction), and poorer charge-transport capability is found in the \( b \)-axis (channel) direction. The inversed \( m^* \) plot of Cy₆–BQQDI shows a more uniform charge-transport behavior, where a favorable charge-transport capability can be found along the channel direction of its OFETs. The high
polycrystalline device performance of Cy₆–BQQDI despite having the orientationally disordered thin-film assembly could be attributed to its isotropic-like charge-transport capability. The current results suggest the importance of molecular design not only in the bulk crystal state, but also in the thin-film state along the channel direction for achieving high device performances.

In summary, the current work reports interesting distinct effects of two sterically demanding substituents on the molecular assemblies in both bulk crystal and thin-film states. Their crystal packing structures lead to different charge-transport capabilities, where the Cy₆–BQQDI derivative exhibits uniform transfer integrals and effective mass compared to the Ph–BQQDI counterpart. Despite having no apparent interlayer interactions, the π-cores of Cy₆–BQQDI do not show significantly larger molecular fluctuations than Ph–BQQDI. In fact, the rigid phenyl groups with multiple interlayer interactions might have led to noticeable fluctuations of the atoms in the bay positions. Though from the tight-binding approximations, the smallest effective mass of Ph–BQQDI is smaller than that of Cy₆–BQQDI, the effective mass of Cy₆–BQQDI along the OFET channel direction is smaller than that of Ph–BQQDI. Thus, OFETs of Cy₆–BQQDI exhibit an excellent μₑ of 2.3 cm² V⁻¹ s⁻¹ in single-crystalline thin film, and up to 1.0 cm² V⁻¹ s⁻¹ in polycrystalline devices. The results herein demonstrate an effective molecular design for molecular assembly, charge transport, and suppressing molecular fluctuations in the bulk single-crystal state, as well as controlling the molecular assembly in the thin-film device state for achieving the optimum charge-transport capabilities. Future work based on the current encouraging results of Cy₆–BQQDI may involve incorporations of alkyl substituents on the cyclohexyl group to improve its solubility for large-area device fabrications.

**Methods**

**Materials and general characterizations.** All amine reagents used in this study were purchased from Tokyo Chemical Industry Co., Ltd and propionic acid was purchased from FUJIFILM Wako Pure Chemical Industries, Ltd without further purifications. o-dichlorobenzene (o-DCB) was purchased from KANTO chemical Co., Ltd. and purified by a solvent purification system. Starting materials 3,4,9,10-benzo[de]isoquinolinof[1,8-gh]quinolinetetracarboxylic dianhydride (BQQ–TCDA) and 3,9-dimethyl-4,10-bis(2,4,6-trichlorophenyl)benzo[de]isoquinolinof[1,8-gh]quinoline-3,4,9,10-tetracarboxylate (BQQ–TCP) were synthesized and purified in our laboratory prior to this study. All reactions were carried out under an atmosphere of argon. ¹H NMR spectra were recorded on JEOL ECS400 spectrometer (400 MHz). Chemical shifts were reported in parts per million (ppm, δ scale).
from residual protons in the deuterated solvent for $^1$H NMR (5.93 ppm for 1,1,2,2-tetrachloroethane-\(d_2\) (TCE-\(d_2\)), 1,1,1,3,3,3-hexafluoro-2-propanol-\(d_2\) (HFIP-\(d_2\)) and 7.26 ppm for chloroform-\(d\) (CDCl$_3$)). The data were presented in the following format: chemical shift, multiplicity (s = singlet, d = doublet, t = triplet, quint = quintet, m = multiplet), coupling constant in Hertz (Hz), signal area integration in natural numbers. Time-of-flight high-resolution mass (TOF-MS) spectrometry measurements were measured on a BRUKER compact-TKP2 mass spectrometer with the atmospheric pressure chemical ionization (APCI) method. Elemental analysis measurements were carried out on a JScience Lab JM10 CHN analyzer at the Comprehensive Analysis Center, the Institute of Scientific and Industrial Research, Osaka University.

**Synthesis.** Detailed synthetic procedures, compound characterizations, and $^1$H NMR of BQQDI derivatives are reported in the Supplementary Information.

**Theoretical calculations.** Estimations of transfer integral and effective mass were conducted using the GAMESS package$^{39}$. The Kohn–Sham eigenstates of all compounds in this work were calculated at the PBEPBE/6-31G(d) level of theory. Transfer integrals between LUMOs of neighboring molecules in the crystal structures were estimated by the dimer method$^1$. To further understand the charge-transport capabilities in the single-crystal state, their LUMO band structures $E(k)$ were calculated by the tight-binding approximation using transfer integrals. Intermolecular interaction energy between two adjacent molecules were obtained at the M06-2X/6-31++G(d,p) level of DFT with counterpoise correction for the basis set superposition error$^{33}$. The calculations were performed using the Gaussian 09 program package$^{40}$.

**X-ray crystallography.** Ph–BQQDI single crystals were obtained by means of PVT, and Cy$_6$–BQQDI crystals were grown in the mixture of nitrobenzene and 1-chloronaphthalene via the slow-cooling method. Single-crystal X-ray diffraction data were collected on a Rigaku R-AXIS RAPID II imaging plate diffractometer with CuK$\alpha$ radiation ($\lambda = 1.54187$ Å) at room temperature. The structures were solved by direct methods [SHELXT (2015)] and refined by full-matrix least-squares procedures on F2 for all reflections [SHELXL (Ver. 2014/7)]. While positions of all hydrogen atoms were calculated geometrically, and refined by applying riding model, all other atoms were refined anisotropically. Crystallographic data have been deposited in the Cambridge Crystallographic Data Centre as a supplementary publication. These data can be obtained free of charge at www.ccdc.cam.ac.uk/data_request/cif.
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\textbf{Author contributions}

T.O. conceived and designed the work, while C.P.Y., N.K., S.K., T.K. synthesized the compounds. C.P.Y., S.K., and N.K. performed the physicochemical properties measurements, single-crystal and thin-film X-ray analyses, and OFET evaluations. H.I. and C.P.Y. calculated the transfer integrals and effective masses. G.W. performed the molecular dynamics simulations. C.P.Y. performed the DFT studies. C.P.Y. and T.O. wrote the manuscript. J.T. and T.O. supervised the work. All authors discussed the results and reviewed the manuscript prior to submission.

\textbf{Competing interests}

The authors declare no competing interests.