Hollow IF-MoS$_2$/r-GO Nanocomposite Filled Polyimide Coating with Improved Mechanical, Thermal and Tribological Properties

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Abstract: Polyimide (PI) is one of the most excellent polymers for coating. However, the high friction coefficient and the high wear rate of pure PI limit its further applications. In this work, the hollow inorganic fullerene-like MoS$_2$/reduced graphene oxide (HIF-MoS$_2$/r-GO) nanocomposite filled PI coating is prepared by in situ polymerization. Reinforcement in mechanical strength and thermal stability is realized on the PI composite coating with incorporation of HIF-MoS$_2$/r-GO, which performs better than carbon nanofiber (CNF). Reduced elastic modulus and hardness of HIF-MoS$_2$/r-GO/PI coating is increased by 8.3% and 4.8%, respectively. The addition of HIF-MoS$_2$/r-GO also results in 24% higher residual mass at 800 °C than CNF. Tribological study indicates that, HIF-MoS$_2$/r-GO/PI coating achieves a wear rate reduction of 79% compared with pure PI under dry sliding condition, which is much more effective than other nanofillers including CNF, r-GO nanosheets and MoS$_2$ nanoparticles. Under liquid lubricated condition, the presence of HIF-MoS$_2$/r-GO in PI results in a 30% reduction in wear rate and 10% reduction in friction coefficient as compared to pure PI. It is thought that the HIF-MoS$_2$/r-GO in PI can be slowly released to the frictional interface and form a protective film during sliding, in this way the aggregation problem is successfully solved.

Keywords: nanocomposite coating; structure; reinforcement; friction and wear; graphene; inorganic fullerene-like MoS$_2$

1. Introduction

Polymers are very promising candidates in a wide range of tribological applications for their excellent mechanical strength, low friction, high chemical stability, light weight, etc. [1–4]. They can not only be used as substitutes for traditional materials, but also as coatings on these materials to improve their lifetime and widen their service conditions [5–7]. Polyimide (PI) is one of the most excellent polymers for coating due to its outstanding thermostability, mechanical properties and chemical stability, especially in severe service conditions [8]. However, the high friction coefficient and the high wear rate of pure PI limit its further applications. In view of this point, great efforts have been made to reinforce PI using functional nanofillers, e.g., SiO$_2$, ZnO, carbon nanotube and graphene [9–12]. The enhancing mechanisms on tribological properties of nanofillers/polymer coatings mainly include higher strength and creep resistance, the formation of stable transfer film giving low friction coefficient, and the micro-rolling/sliding effect of nanoparticles [13–15].

In recent years, graphene and MoS$_2$-based materials have proven to be very effective additives in polymers. Graphene and their derivatives can realize the high strength as well as high tribological properties in very low content [8,16,17]. MoS$_2$ is one of the most prominent lubricating fillers with ultra-high load-carrying capability and excellent friction-reducing performance. Until now, MoS$_2$ in various sizes and morphologies have been...
extensively studied as fillers in polymers [18,19]. It is widely recognized that the family of two-dimensional (2D) nanomaterials can enter the frictional interface and reduce direct contact between the polymer and counterpart. Yuan et al. [19] prepared the 2D MoS$_2$ nanosheets by a high shear mixing technique and filled them into polyamic acid (PAA) precursors to get the MoS$_2$/PI nanocomposite film. The reported MoS$_2$/PI film showed significantly enhanced mechanical and thermal properties.

In addition, the properties of polymer composite coatings are greatly affected by the nanofiller’s shape and the way it is assembled [14]. Due to the high dispersion and special microstructure, nanocomposite can remarkably improve the mechanical and tribological properties of polymers. Yuan et al. [20] reported a facile and efficient approach to overcome the poor dispersion of MoS$_2$ nanoflowers in PI by carefully grafting them onto the surface of hollow carbon nanofibers (HNCNF). It revealed that tribological properties of PI composite coatings were effectively improved under different lubrication conditions after incorporating small amounts of MoS$_2$/HNCNF. Xin et al. [11,21] prepared the multi-dimensionally assembled nanocomposites of modified graphene/carbon nanotube (CNT) and MoS$_2$/CNT by chemical interaction; both of them greatly increased the mechanical strength and tribological properties of PI-based coatings. These works indicate that, as multidimensional nanohybrids, the geometric microstructure has a crucial influence on achieving optimal performance. Also based on the impact of structure, we recently synthesized a novel hollow fullerene-like MoS$_2$/reduced graphene oxide (HIF-MoS$_2$/r-GO) nanocomposite by the hydrothermal method [22]. As a filler in the ionic grease, HIF-MoS$_2$/r-GO showed a great advantage in lubricating properties over other additives including MoS$_2$ nanoparticles, and single and physically mixed forms of r-GO or HIF-MoS$_2$. The application of HIF-MoS$_2$/r-GO nanocomposite should be far beyond this. In consideration of the sliding/rolling of spherical HIF-MoS$_2$ at the interface [23], it is very likely that HIF-MoS$_2$/r-GO will be an excellent additive in polymer composite coatings.

From another point of view, the wear rate and friction coefficient of polymers can be remarkably reduced with liquid lubrication [24–27]. Moreover, adding fillers in either polymers or lubricants can further enhance tribological properties of the lubricated polymers. The work by Zhao et al. [28] showed that the addition of short glass fibers (SGF) and ZnS particles into PI led to a synergetic role in friction reduction. Additionally, under low sliding speeds with the limited hydrodynamic effect of oil film, ZnS/SGF/PI exhibited enhanced wear resistance in comparison to the composite reinforced only with SGF. In our previous work, graphene oxide (GO) was added to polyetheretherketone (PEEK) or water to improve the tribological properties of water-lubricated PEEK-steel [29]. Results indicated that in comparison with adding GO into PEEK, adding GO into water exhibits much better friction-reducing and anti-wear properties. However, different from adding fillers into the polymers, it is normally difficult to get the well dispersed lubricants containing inorganic nanoparticles such as graphene and MoS$_2$. Surface functionalization with surfactant-like substances has been widely studied to improve the dispersity, but the modified nanoparticles are prone to degradation under tribological conditions leading to unstable lubricating properties [30].

In this work, the HIF-MoS$_2$/r-GO filled PI coating was prepared by in situ polymerization. The mechanical properties, thermal stability and tribological properties of the coatings were studied. Carbon nanofiber (CNF), which has proven to be as an effective additive for polymers [31,32], was studied for comparison. Additionally, the commercial MoS$_2$ nanoparticles and pure r-GO were filled into the PI coating respectively for tribological tests. Furthermore, ionic liquid (IL) was used as a liquid lubricant to study the effect of HIF-MoS$_2$/r-GO in PI under the lubricated condition. It was expected to achieve a synergistic effect on the PI coating by using liquid lubricant and incorporating HIF-MoS$_2$/r-GO nanocomposite.
2. Materials and Methods

2.1. Materials

Bisphenol-A dianhydride (BPADA, 99% purity, Shandong XiYa Chemical Industry Co., Ltd., Jinan, China), 4,4′-oxydianiline (ODA, 98% purity, Meryer (Shanghai, China) Chemical Technology Co., Ltd., Shanghai, China), and N,N-Dimethylformamide (DMF, 99.5% purity, Sinopharm Chemical Reagent Co., Ltd., Shanghai, China) were used as received. CNF was purchased from Beijing Deke Island Gold Technology Co., Ltd., Beijing, China. Ionic liquid of 1-ethyl-3-methylimidazolium bis(trifluoromethyl sulfonyl)imide ([Emim][NTf$_2$], 99% purity) was supplied by Linzhou Keneng Materials Technology Go., Ltd., Anyang, China.

2.2. Preparation of Additives and PI Composite Coatings

HIF-MoS$_2$/r-GO nanocomposite was synthesized by the hydrothermal method; the detailed process was included in our previous work [22]. GO was synthesized by the improved Hummers’ method described in the previous paper [33]. r-GO was fabricated by thermally reducing GO. The thermal reduction condition was the same as that in HIF-MoS$_2$/r-GO synthesis, that is, GO was heated to 800 °C at 5 °C/min under Ar flowing for 2 h.

The PI composite coatings were prepared on polished 1045 steel substrates, and the preparation included in situ polymerization, tape casting and thermal imidization processes. In a typical operation to prepare the HIF-MoS$_2$/r-GO filled PI coating, ODA (2.503 g, 12.5 mmol) was added into a 50 mL two-necked flask containing 38 mL DMF with aid of sonication for 10 min, then the as-prepared HIF-MoS$_2$/r-GO at different loadings (i.e., 0.5%, 1.0%, 1.5% and 2% by weight) were added into the above solutions and dispersed by sonication until the nanocomposites were dispersed homogeneously in the mixed solution. BPADA (6.506 g, 12.5 mmol) was gradually added into above mixed solution, and the mixture was stirred at room temperature for 24 h under nitrogen. Finally, the HIF-MoS$_2$/r-GO/polyamic acid (PAA) solutions were obtained via in situ polymerization. The as-formed HIF-MoS$_2$/r-GO/PAA solutions were dropped onto polished steel blocks. Subsequently, the mixtures were dried in a vacuum oven at 80 °C for 10 h by evaporating solvent. The thermal imidization process was performed in a muffle furnace at 70 °C for 2 h, and at 100, 150, 200 and 300 °C for 1 h. The other coatings were also prepared by the above-mentioned method. The coating thickness was around 50 µm.

2.3. Characterization of Coatings

The morphology and composition of the HIF-MoS$_2$/r-GO nanocomposite were studied in our previous work, using high-resolution transmission electron microscopy (TEM) with an energy dispersive X-ray spectroscopic (EDX) detector, Raman spectroscopy, and a powder X-ray diffractometer (PXRD) [22]. To check the distribution of fillers in PI composite coatings, the coatings were fractured to examine the microscopic cross section with a scanning electron microscopy (SEM, Hitachi S-4800, Tokyo, Japan). The coatings for cross section study and thermogravimetric analysis (TGA) were prepared on silica glass in convenience to be peeled off. TGA was performed by the Discovery SDT 650 analyser (TA Instruments, Inc., New Castle, DE, USA) from 30 to 850 °C in a nitrogen (N$_2$) atmosphere at a heating rate of 10 °C/min. Three repeated TGA tests were done to minimize the error range. Mechanical properties of PI-based coatings were evaluated by a nanoindenter (Agilent G200, Santa Clara, CA, USA) with following parameters: peak/maximum indentation load of 30 mN with a loading and unloading rate of 0.05 mN/s, and a holding time of 50 s at the peak loads. The mean values of hardness (H) and reduced elastic modulus (E$_r$) were obtained by averaging at least three different surface positions for each sample.

2.4. Tribological Tests

A multifunctional material surface performance tester (model CFT-I, Lanzhou Zhongke Kaihua Technology Development Co., Lanzhou, China) operated by ball-on-disc mode...
was used to evaluate the friction and wear behaviours of PI-based coatings. Commercially available 100Cr6 steel ball (diameter 6 mm, surface roughness Ra ≤ 30 nm) was used as the stationary upper counterpart, while the PI-based coatings were mounted onto the flat base. The flat base was driven to reciprocally move at a frequency of 10 Hz with a stroke length of 5 mm. For tribological properties under dry sliding condition, tests were conducted at room temperature of 25 ± 5 °C, elevated temperatures of 100 and 250 °C, and the relative humidity was 50 ± 5%. The applied loads were 2 and 6 N for the room temperature test and 6 N for the high temperature test; the testing duration was 30 min. For tribological properties under the lubricated condition, [Emim][NTf₂] was dropped on the coatings, forming an IL-lubricated condition. Due to the negligible wear at 2 and 6 N under the IL-lubricated condition, the applied load was increased to 20 N to evaluate the tribological properties.

The friction coefficient-versus-time curve was recorded automatically with a data acquiring system linked to the tester. Wear volume (V) of the coatings was tested by scanning the wear scars and accumulating the volume loss. The specific wear rate K (mm³/(N·m)) was calculated by Equation (1).

\[ K = VF^{-1}L^{-1} \]  

where F and L are the normal load and sliding distance of the wear test, respectively. At least three repeated measurements were performed for each frictional pair. The worn surface of coatings generated from PI, CNF/PI and HIF-MoS₂/r-GO/PI under dry sliding condition was studied using an optical microscope (Model MV6100, Nanjing Jiangnan Novel Optics Co., Ltd., Nanjing, China) and SEM.

3. Results and Discussion

3.1. Mechanical and Thermal Properties

Figure 1 shows the SEM images of the fractured surface for different PI coatings. Inserts in Figure 1b,c are the as-prepared HIF-MoS₂/r-GO and as-received CNF, respectively. Pure PI coating has a smooth fractured surface (Figure 1a), and the CNF-reinforced coating also shows a smooth surface with incorporated CNF (Figure 1c). On the contrary, HIF-MoS₂/r-GO/PI has an irregular fractured surface where the added HIF-MoS₂/r-GO can be observed (Figure 1b). On the one hand, it reveals that the added HIF-MoS₂/r-GO nanocomposite is well dispersed in PI. On the other hand, the ball-on-plate structure of HIF-MoS₂/r-GO may help form a robust barrier throughout the nanocomposite coating.

Figure 1. Scanning electron microscopy (SEM) images of fractured surface for different polyimide (PI) composite coatings: (a) PI, (b) 1.5%HIF-MoS₂/r-GO/PI and (c) 1.5%CNF/PI. Inserts in (b) and (c) are the HIF-MoS₂/r-GO and carbon nanofiber (CNF) at the same magnification, respectively. 1.5%HIF-MoS₂/r-GO/PI = hollow inorganic fullerene-like MoS₂/reduced graphene oxide.

The reduced elastic modulus (E_r) and hardness (H) of different PI coatings are compared in Figures 2a,b, respectively. It is found that both E_r and H are improved after the addition of HIF-MoS₂/r-GO and CNF. E_r and H of pure PI is 3.97 GPa and 0.165 GPa. As the content of HIF-MoS₂/r-GO reaches 1.5%, the E_r and H of HIF-MoS₂/r-GO/PI is increased to 4.30 GPa and 0.173 GPa, with increasing rates of 8.3% and 4.8%, respectively. While
the presence of 1.5%CNF increases $E_r$ and $H$ by 7.3% and 4.2%. It is noteworthy that the 2%HIF-MoS$_2$/r-GO/PI and 2%CNF/PI coatings show rough surface with some cracks and holes on the surface, which should be caused by the too high loading of HIF-MoS$_2$/r-GO. As the content of nanofillers is more than 1.5%, the viscosity of solution obviously increases, which affects the coating preparation. Moreover, the nanofillers would even aggregate and lead to the poor dispersion. Because of this, the mechanical properties of them were not studied by nanoindenter.

TGA was carried out to investigate the influence of nanofillers on thermal stability. Although the TGA study does not directly examine the filler influence on the performance of PI nanocomposite at its working temperature, it may shed light on the micro interaction between the fillers and PI, which would affect the PI nanocomposite’s performance. Based on the thermogravimetric (TG) curves shown in Figure 3, the decomposition temperature, the temperature at 5% and 10% weight loss, and the residual amount at 800 °C are obtained. The mean value and standard deviation are listed in Table 1. The results indicate that HIF-MoS$_2$/r-GO and CNF can improve the thermal stability of PI coating, with the optimum loading of 1.5%. The residual amount at 800 °C for pure PI and 1.5%HIF-MoS$_2$/r-GO/PI is 47.5% and 65.0%, respectively. The residual amount for 1.5%HIF-MoS$_2$/r-GO/PI is 37% higher than pure PI. 2.0%HIF-MoS$_2$/r-GO/PI shows relatively lower thermal properties than 1.5%HIF-MoS$_2$/r-GO/PI, which should be induced by the defects in the PI composite with too high loading of HIF-MoS$_2$/r-GO. The thermal stability of HIF-MoS$_2$/r-GO/PI is higher than that of CNF/PI. The addition of 1.5%HIF-MoS$_2$/r-GO results in 24% higher residual amount at 800 °C than that of 1.5%CNF. The higher stability of HIF-MoS$_2$/r-GO/PI is thought to be ascribed to improvement of the physical barrier effect by HIF-MoS$_2$/r-GO as compared to CNF, which could delay the thermal degradation of PI [34]. During thermal degradation, small molecules will emit from the heated samples. Due to the barrier effect by highly thermal stable 2D fillers, the emission can be inhibited, and the thermal degradation will be delayed. Since CNF is in fibre shape, it provides very limited area to protect PI. It is worth mentioning that the PI composite should have higher thermal properties than the mixture obtained via physical mixing the components, because the PAA precursor is in situ polymerized on the surface of nanocomposite and the produced PI highly adheres to the surface. It can be revealed from the result that very low content of fillers (≤2 wt.%) in PI composites induces a big difference in the residual amount. The higher thermal properties of HIF-MoS$_2$/r-GO/PI also suggest that after adding the HIF-MoS$_2$/r-GO nanocomposite, a robust protective barrier could be formed within the nanocomposite coating.

Figure 2. (a) The reduced elastic modulus ($E_r$) and (b) the hardness ($H$) of different PI composite coatings.
Table 1. Thermal analysis of different PI composite coatings.

| PI Coatings       | $T_{\text{onset}}$ (°C) | $T_{d5}$ (°C) | $T_{d10}$ (°C) | Residual Amount at 800 °C (%) |
|-------------------|--------------------------|--------------|----------------|-------------------------------|
| PI                | 506.6 ± 2.2              | 505.6 ± 1.8  | 520.4 ± 1.5    | 47.5 ± 1.6                    |
| 0.5%HIF-MoS$_2$/r-GO/PI | 510.0 ± 0.4              | 516.7 ± 1.8  | 528.0 ± 1.5    | 58.6 ± 2.0                    |
| 1.0%HIF-MoS$_2$/r-GO/PI | 515.6 ± 0.3              | 524.9 ± 0.5  | 535.7 ± 0.8    | 63.5 ± 0.8                    |
| 1.5%HIF-MoS$_2$/r-GO/PI | 519.3 ± 0.5              | 527.2 ± 1.1  | 536.7 ± 1.2    | 65.0 ± 2.1                    |
| 2.0%HIF-MoS$_2$/r-GO/PI | 514.1 ± 0.4              | 524.0 ± 0.7  | 534.7 ± 0.4    | 61.0 ± 1.2                    |
| 0.5%CNF/PI        | 506.5 ± 1.9              | 507.7 ± 1.9  | 520.6 ± 1.7    | 51.8 ± 0.2                    |
| 1.0%CNF/PI        | 506.3 ± 0.7              | 507.3 ± 0.6  | 519.7 ± 0.7    | 51.8 ± 1.4                    |
| 1.5%CNF/PI        | 508.2 ± 2.0              | 508.1 ± 1.3  | 521.4 ± 0.3    | 52.0 ± 1.5                    |
| 2.0%CNF/PI        | 508.0 ± 1.5              | 507.2 ± 2.0  | 521.0 ± 0.6    | 49.7 ± 1.4                    |

Note: (a) Initial decomposition temperature; (b) Temperature at 5% weight loss; (c) Temperature at 10% weight loss.

3.2. Tribological Properties

3.2.1. Under Dry Sliding Condition

Tribological properties of different PI composite coatings at dry sliding condition and room temperature are compared in Figure 4. As shown in Figure 4a, the friction coefficient of pure PI is 0.492. After adding 1% HIF-MoS$_2$/r-GO, the friction coefficient is reduced to 0.431, with a reduction of 12%. While CNF can maximally reduce the friction coefficient by just 6% at 0.5% loading. Higher CNF even leads to an increase in friction coefficient as compared to pure PI. The anti-wear property of PI composite coating is remarkably improved with HIF-MoS$_2$/r-GO (Figure 4b). At the optimum loading of 1.5%, the wear rate can be reduced by 79%, which is much better than the other fillers at their optimum content. It is noteworthy that the improvement by adding HIF-MoS$_2$/r-GO is much greater than just adding r-GO or MoS$_2$ nanoparticles.
The reinforcing effect of HIF-MoS$_2$/r-GO and CNF at their optimum loading, i.e., 1.5%, was further examined under a higher load of 6 N at room temperature. Pure PI and 1.5%CNF/PI both show increased friction coefficient over time, while 1.5%HIF-MoS$_2$/r-GO/PI shows lower and stable friction coefficient throughout the test run (Figure 5). The main reason for the increased friction coefficient is thought to be the increased contact surface for the poor anti-wear property. Pure PI is even worn out and the steel surface is exposed (Figure 6a,d). 1.5%CNF/PI shows improved anti-wear performance with a wear depth of 25 μm (Figure 6f). By filling 1.5%HIF-MoS$_2$/r-GO, the wear depth is just around 13 μm (Figure 6e). The worn surface of 1.5%HIF-MoS$_2$/r-GO/PI (Figure 6b) is smoother compared to pure PI (Figure 6a) and 1.5%CNF/PI (Figure 6c). Under higher load, the added HIF-MoS$_2$/r-GO is still able to strengthen the anti-wear transfer film and improve the tribological properties of PI.

**Figure 4.** The effect of filler type and content on (a) friction coefficient and (b) wear rate of PI composite coatings under dry sliding condition at room temperature (Load: 2 N).

**Figure 5.** The effect of different fillers on friction coefficient of PI composite coatings under dry sliding condition at room temperature (Load: 6 N).
Figure 6. Optical micrographs of worn surfaces of PI coatings with different fillers under dry sliding condition at room temperature: (a) PI, (b) 1.5%HIF-MoS$_2$/r-GO/PI, (c) 1.5%CNF/PI; (d–f) are worn surface profiles corresponded to (a–c) (Load: 6 N).

Tribological properties of 1.5%HIF-MoS$_2$/r-GO/PI were further studied at elevated temperatures of 100 and 250 °C, and the results of wear rate and online friction coefficient are depicted in Figures 7a and 7b, respectively. At 100 °C, the wear rates of both pure PI and 1.5%HIF-MoS$_2$/r-GO/PI coatings are much lower than those at room temperature. At high temperature, more serious adhesive wear is induced instead of abrasive wear. The softened PI material is easier to be sheared to form an anti-wear film at the interface. Compared to the pure PI coating, the wear rate of 1.5%HIF-MoS$_2$/r-GO/PI can be reduced by 32% and 53% at 100 °C and 250 °C, respectively. At a higher temperature of 250 °C, the wear rates are much higher than 100 °C, which should be caused by the reduced mechanical properties [35]. Pure PI with too poor strength cannot stand the shearing force, as a result the coating at 250 °C shows higher wear than 100 °C. The friction coefficient shows an increasing trend as the temperature increases from 100 °C to 250 °C. It is thought to be induced by the higher adhesion and higher real contact surface.

Figure 7. (a) The wear rate and (b) online friction coefficient of pure PI and 1.5%HIF-MoS$_2$/r-GO/PI coatings under dry sliding condition at 100 and 250 °C (Load: 6 N).

To shed light on the mechanism of different tribological properties, the worn surface of the PI-based coatings and wear debris on the steel ball were studied by SEM. Pure PI as
shown in Figure 8a shows wide and deep furrows along the sliding direction. Figure 8d shows the SEM image of wear debris on the steel ball sliding against pure PI coating. Many stacked wear debris are found with rough edges. These large and thick wear debris indicate that pure PI is easy to wear out, as a result the steel ball slides directly against the fresh PI and leads to a high wear rate. After filling HIF-MoS$_2$/r-GO, the worn surface becomes much smoother with some adhered thin wear debris (Figure 8b). There is much less wear debris around the worn surface and the steel ball. From the SEM image of Figure 8e, we find that the wear debris of HIF-MoS$_2$/r-GO/PI is relatively thinner and smoother than that of pure PI. In the magnified images, there are small particles of wear debris on the surface for HIF-MoS$_2$/r-GO/PI (Figure 8h), while there are very few particles for pure PI (Figure 8g). Figure 8i displays the enlarged region within the rectangular frame, where HIF-MoS$_2$ nanocages can be found on the surface. The detected HIF-MoS$_2$ nanocages indicate that they can be released from the HIF-MoS$_2$/r-GO nanocomposite during the frictional process. On one hand, the added HIF-MoS$_2$/r-GO nanocomposite and HIF-MoS$_2$ on the surface of r-GO can be released to the friction interface to reduce direct contact between steel ball and PI coating. On the other hand, they can in situ repair the worn surface and help form thinner but robust wear debris at the interface to prevent the coating from being further damaged. By filling CNF, the worn surface has deep furrows and microcracks (Figure 8c), and the wear debris as shown in Figure 8f is similar to that of pure PI coating. The lower tribological properties of CNF/PI compared with HIF-MoS$_2$/r-GO/PI should be due to the lower ability of CNF to form the transfer film.

**Figure 8.** SEM images of (a–c) worn surfaces and (d–i) wear debris on the steel ball for different PI coatings under dry sliding condition at room temperature: (a,d) PI, (b,e) 1.5%HIF-MoS$_2$/r-GO/PI, and (c,f) 1.5%CNF/PI. Insert in (d) shows the overall image of wear debris on steel ball sliding against pure PI coating. The magnified SEM images of wear debris for PI and HIF-MoS$_2$/r-GO/PI are shown in (g) and (h), respectively, and (i) displays the enlarged region within the white rectangular frame in (h). (Load: 2 N).
The structure of HIF-MoS$_2$/r-GO filled PAA precursors is shown in Figure 9a, and the proposed friction and wear mechanism of HIF-MoS$_2$/r-GO/PI coating is shown in Figure 9b. Different from the fibrous CNF, the HIF-MoS$_2$ on r-GO can act as nano balls after being peeled from the r-GO surface (Figure 9b). Additionally, after the breakage of HIF-MoS$_2$ nanocages, the MoS$_2$ nanosheet debris could also fill the worn surface to form the anti-wear and friction-reducing transfer film. Regarding the r-GO nanosheets, they can not only work as a robust anti-wear additive, but also serve as the storer for HIF-MoS$_2$ for slow release of HIF-MoS$_2$. Aggregation is a common problem in the preparation and performing of nanoparticles filled materials. By assembling HIF-MoS$_2$ and r-GO, the aggregation problem can be avoided. The nano-sized HIF-MoS$_2$ dotted on r-GO can avoid the aggregation of r-GO nanosheets. During the preparation of HIF-MoS$_2$/r-GO/PI composite coating, the PI can fully adhere to HIF-MoS$_2$ and r-GO because of the enlarged gaps between HIF-MoS$_2$/r-GO nanocomposites, which can not only enhance the mechanical properties, but also realize the slow release of nanoparticles under shearing force.

![Figure 9.](image_url) (a) The structure of HIF-MoS$_2$/r-GO filled PAA precursors, and (b) the proposed friction and wear mechanism of HIF-MoS$_2$/r-GO/PI under dry sliding condition.

3.2.2. Under Lubricated Condition

For the IL-lubricated condition, commercial IL of [Emim][NTf$_2$] was dropped on the coating as a lubricant. The PI coating shows much lower wear and friction coefficient under IL-lubricated condition as compared to the dry sliding condition (Figure 10). It is exciting that the addition of HIF-MoS$_2$/r-GO results in a 30% reduction of wear as compared to pure PI, and the friction coefficient is further reduced by 10%. It is noteworthy that tribological tests were also conducted under low loads of 2 and 6 N, but the wear of coatings was too low to be detected by the machine. Even under 20 N, the wear rate of PI and HIF-MoS$_2$/r-GO/PI coating is just $10 \times 10^{-7}$ and $7 \times 10^{-7}$ mm$^3$N$^{-1}$m$^{-1}$, respectively.

![Figure 10.](image_url) The wear rate and friction coefficient of IL-lubricated PI and 1.5%HIF-MoS$_2$/r-GO/PI (Load: 20 N).

Under the testing conditions, the IL-lubricated contact is in the mixed regime of elastohydrodynamic lubrication and boundary lubrication. From the mechanical perspective,
the filled HIF-MoS\textsubscript{2}/r-GO enhances the mechanical strength of the PI coating, giving rise to a higher anti-wear property. From the microscale interfacial effect, HIF-MoS\textsubscript{2}/r-GO and the deciduous HIF-MoS\textsubscript{2} can be slowly released from the coating to the frictional interface, which can prevent direct contact of the PI coating and the steel ball. The lower friction coefficient indicates that the HIF-MoS\textsubscript{2}/r-GO in PI can in situ form a protective tribological film during sliding. Due to the ball-on-plate structure of HIF-MoS\textsubscript{2}/r-GO nanocomposite, the released HIF-MoS\textsubscript{2} acts as rolling balls which can prevent the contact between the asperities of two sliding surfaces, resulting in a further decrease in the friction coefficient. In case of a severe contact condition where the HIF-MoS\textsubscript{2} could be broken, the thin IF-MoS\textsubscript{2} nanosheets could also reduce the friction and wear.

4. Conclusions

The HIF-MoS\textsubscript{2}/r-GO nanocomposite is directly filled into the PI coating during the preparation of polyamic acid precursors, followed by in situ polymerization and thermal imidization. Reinforcement in mechanical strength, thermal stability and tribological properties are realised on the HIF-MoS\textsubscript{2}/r-GO/PI nanocomposite coating. Reduced elastic modulus and hardness of HIF-MoS\textsubscript{2}/r-GO/PI coating is increased by 8.3% and 4.8%, respectively. The addition of HIF-MoS\textsubscript{2}/r-GO in PI also results in 24% higher residual mass at 800 °C as compared to CNF. SEM images suggest that the ball-on-plate structure of HIF-MoS\textsubscript{2}/r-GO may help form a robust protective barrier throughout the nanocomposite coating.

In terms of tribological properties, HIF-MoS\textsubscript{2}/r-GO achieves a wear rate reduction of 79% under dry sliding condition, which is much more effective than the other additives, including r-GO, MoS\textsubscript{2} nanoparticles and CNF. Moreover, HIF-MoS\textsubscript{2}/r-GO still performs well at high temperature of 250 °C, which can reduce the wear rate of PI coating by 53%. Under the IL-lubricated condition, the PI coating shows much lower wear and friction coefficient than the dry sliding condition. It is exciting that adding HIF-MoS\textsubscript{2}/r-GO results in 30% reduction in wear rate and a 10% reduction in the friction coefficient. It indicates that the HIF-MoS\textsubscript{2}/r-GO in PI can in situ form a protective film during sliding; in this way the aggregation problem is successfully solved.

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