ABSTRACT: The individual and collective behavior of ions near electrically charged interfaces is foundational to a variety of electrochemical phenomena encountered in biology, energy, and the environment. While many theories have been developed to predict the interfacial arrangements of counterions, direct experimental observations and validations have remained elusive. Utilizing cryo-electron microscopy, here we directly visualize individual counterions and reveal their discrete interfacial layering. Comparison with simulations suggests the strong effects of finite ionic size and electrostatic interactions. We also uncover correlated ionic structures under extreme confinement, with the channel widths approaching the ionic diameter (∼1 nm). Our work reveals the roles of ionic size, valency, and confinement in determining the structures of liquid–solid interfaces and opens up new opportunities to study such systems at the single-ion level.

KEYWORDS: electrical double layer, cryo-electron microscopy, single-ion imaging, ionic distribution
Figure 1b shows a representative cryo-TEM image of the interface between a positively charged nanorod and the electrolyte solution. In general, such interfaces can be divided into two characteristic regions. At the nanorod surface, the Keggin anions accumulate and form a dense and ordered structure. Beyond this dense structure, a homogeneous bulk solution appears, in which the Keggin anions are more sparsely distributed. To reveal the ionic distribution perpendicular to the interface, we integrate the image intensity along the length of the nanorod (y-axis), as shown in Figure 1c. The emergence of four distinct peaks along the x-axis indicates that the Keggin anions form four discrete layers at the interface, with an averaged separation of 1.3 nm, which is slightly larger than the diameter of a dehydrated Keggin anion (1.1 nm). The prominence of the first peak suggests that the Keggin anions are densely packed in the first layer next to the surface.

We further study the spatial ionic ordering along the interface (y-axis) by performing the fast Fourier transform (FFT) of the image profile for each layer of the Keggin anions. We observe an FFT peak in all of the four layers at the interface, in sharp contrast to a region in the bulk solution, which does not show an FFT peak (see Figure 1d). In particular, all of the peak positions are in the same spatial frequency range of 0.6–0.7 nm⁻¹, which corresponds to the spatial periodicity range of 1.4–1.7 nm. We note that such spatial periodicity along the y-axis is slightly larger than the averaged periodicity of 1.3 nm along the x-axis. This asymmetry suggests that the Keggin anions are pushed onto the surface of the nanorod because of their attractive interaction with the surface charge. In addition, the FFT peaks become broadened from the first layer to the fourth layer, indicating that the ionic distribution becomes less ordered with increasing distance from the charged surface.

The layered distribution of the Keggin anions can be generally explained as an interplay between the finite ion size, the high surface charge density, and the high ion valency. We perform molecular dynamics simulation in order to understand the ionic structure. To capture the essence of the physical mechanism, we represent the problem with a primitive model, where the solvent is described as a structureless continuum with a fixed dielectric constant, and the ions interact via electrostatic and hard-sphere interactions. The curvature of the nanorods is neglected in the simulations (see Supporting Information for details). In the simulation, the effective diameters of the Keggin ions (1.5 nm) include the first hydration shell, and the monovalent counterions (Cl⁻) are excluded. With an experimentally relevant surface charge density of 4.5 e/nm² (ref 50), the simulation shows four layers of the multivalent counterions at the planar interface (Figure 2a), while the hydronium co-ions are repelled from the layering structure (Figure S2). The simulated concentration distribution of the Keggin anions is plotted in Figure 2b. The peak positions of the concentration profile agree well with the peak positions of the integrated image intensity profile (see arrows in Figure 1c). Such an oscillatory concentration profile of multivalent counterions cannot be described by the classical mean-field Poisson–Boltzmann theory (inset of Figure 2b). In order to theoretically capture the layering behavior, the excluded volume of the ions must be considered in their chemical potential, either in terms of local ionic densities or integral-weighted ionic densities as shown by the inset of Figure 2b.

Figure 1. Cryo-TEM imaging of discrete ions and their distributions at the liquid–solid interface. (a) The sample consists of amine-functionalized gold nanorods embedded in an aqueous solution of phosphotungstic acid (H₃PW₁₂O₄₀). The amine functionalization groups are positively charged, which attract and redistribute the Keggin anions [PW₁₂O₄₀]³⁻. A parallel electron beam is incident into the top surface (+z), and the exit wave produces a projected image in the (x−y) plane. (b) A typical cryo-TEM image of the interface between a nanorod and the electrolyte solution (dashed region in panel a). The image contrast in the electrolyte solution is dominated by the Keggin anions, which contain multiple high-Z tungsten atoms. The inset shows a noise-filtered image of a single Keggin anion in vitrified solution. The central positions of the high-intensity spots in the red region are marked on the right. (c) Integrated image intensity profile within 16 nm from the surface of the nanorod (red region in panel b). The oscillatory behavior below 5 nm suggests that the Keggin anions form discrete layers at the interface. The arrows pointing down mark the positions of concentration peaks of the Keggin anions obtained from the molecular dynamics simulation. (d) Fast Fourier transform (FFT) profiles along the y-direction of the image intensity for different layers of the Keggin anions (i−iv marked in panel c). The peak features suggest spatial ionic ordering along the interface. For comparison, the FFT for a region in the bulk solution is plotted (v marked in panel c), which does not show a peak feature. All curves are normalized and vertically offset. The shaded areas represent the confidence bounds of the FFT curves (one standard deviation). The arrows mark the positions of the radially averaged FFT peaks for each layer of the Keggin anions obtained from the molecular dynamics simulation.

microscope (TEM). The pH of the solution was adjusted to 1.2 ± 0.2 by hydrochloric acid (HCl) to stabilize the Keggin anions. After mixing, a few microliters of the solution were applied to a lacey carbon TEM grid, and the excess liquid was blotted away from both sides with filter papers. The grid was then rapidly cooled to 90 K inside liquid ethane for vitrifaction to occur. Finally, samples were transferred into a TEM and imaged under a low electron dose (~100 e/A²) at 100 K to reduce electron beam damage. With an under-defocus of ~250 nm, we obtained adequate image contrast that resolves individual Keggin anions in the vitrified solution (see Supporting Information for details).
We now apply the simulation results to examine the ionic distributions within each layer. Because of the reduction of the electrostatic interaction with the surface, the ionic structures become less ordered from the first layer to the fourth layer, as revealed by the two-dimensional cross-sectional maps of the multivalent counterions (Figure 2c). Within the simulation size (lateral dimension $15 \times 15$ nm$^2$), the ionic distributions in the first two layers possess a hexagonal symmetry, as shown by the two-dimensional FFT profiles (Figure 2d). On a larger scale beyond the simulation, polycrystalline structures could emerge,
and therefore, the two-dimensional FFT profiles would become isotropic. For this reason, we perform radial averaging on the two-dimensional FFTs to represent the ionic ordering along the interface for each layer of the Keggin anions (Figure 2e). The peak positions of the radially averaged FFTs agree well with the peak positions of the FFTs of the image intensity profiles (see arrows in Figure 1d). The above simulation captures the room-temperature interfacial ionic arrangements controlled by electrostatic and excluded volume interactions, supporting the main observations from our cryo-TEM imaging.

The cryo-imaging technique also allows us to visualize ionic distributions under confinement. In the solution, the multivalent anions mediate an “anomalous” attraction between the positively charged surfaces (so-called like-charge attraction),16,51 which leads to aggregation of the nanorods (Figure S3). Such aggregation gives rise to extreme confinement of the electrolyte solution, as shown by the cryo-TEM images in Figure 3a,b. With different confinement widths of 1.3 and 2.5 nm, the Keggin anions form monolayer and bilayer structures between the parallel nanorods, as confirmed by the integrated image intensity profiles in Figure 3c. A variation of the number of layers is also observed in unconfined configuration (Figure S4), suggesting a variation of surface charge density in the sample. The FFT profiles of the image intensity for each layer are plotted in Figure 3d. For all layers, an FFT peak emerges in the spatial frequency range 0.7−0.8 nm−1, corresponding to a spatial periodicity of 1.2−1.4 nm. The observed monolayer and bilayer correlated ionic structures resemble a series of repulsively interacting systems under confinement, such as colloids52 and dusty plasmas.53

In cryo-TEM, the sensitivity of vitrified water to an electron beam restricts the electron dose to ∼100 e/Å2, which limits the imaging spatial resolution by electron shot noise.54 On the basis of the signal-to-noise analysis of our experiment with Keggin ions, we estimate that the smallest diameter of a heavy metal particle that can be resolved is ~0.7 nm (Figure S5). In biological samples, this limitation can be overcome by aligning and averaging numerous low dose images of highly similar biomolecules. Such an averaging method has resolved individual monatomic ions that bind to biological channels.55,56

Another question regarding cryo-TEM is whether the vitrification preserves the intrinsic ionic arrangements. From a macroscopic view of heat and mass transfer, the thermal diffusivities in gold and water are over 2 orders of magnitude higher than the ion diffusivity in water, suggesting that freezing occurs much faster than ionic motion. This is supported by the homogeneity of vitrified ionic solutions, which are supersaturated at cryogenic temperatures17 (Figure S6). Moreover, based on previous research in biology, molecular structures can be preserved by vitrification down to the atomic scale.31−34 This fact is consistent with the compatibility between our experimental observation and the theoretical description of room-temperature electrical double layers.

In conclusion, we have presented, for the first time, cryo-TEM images of discrete counterions in electrical double layers. The distribution of the multivalent Keggin anions at the highly charged surface exhibits discrete layering and strong three-dimensional ordering. The utility of cryo-TEM provides new opportunities to explore the liquid−solid interfaces at the single-ion level. Furthermore, it offers a mechanism to validate and improve theoretical models. Finally, the ability to resolve ionic structures under extreme confinement provides a new route to explore nanofluidic devices and nanostructured electrochemical materials.

- **ASSOCIATED CONTENT**

- **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.nanolett.0c02669.

Sample preparation; cryo-TEM imaging method and signal-to-noise analysis; molecular dynamics simulation; continuum theories; aggregation of amine-functionalized nanorods (PDF)

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H.-K.L., Z.Z., J.C., and A.M. conceived the project. H.-K.L. conducted the experiments. Z.Z., J.M., K.S., and J.C. assisted the experiments. J.P.D. and M.B. performed the theoretical analysis. H.-K.L., J.P.D., and A.M. wrote the manuscript with inputs from all authors. H.-K.L. and J.P.D. contributed equally to this work.

**Notes**

The authors declare no competing financial interest.

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REFERENCES

(1) Hückel, E.; Debye, P. The theory of electrolytes: I. lowering of freezing point and related phenomena. Phys. Z. 1923, 24, 185–206.
(2) Israelachvili, J. N. Intermolecular and surface forces; Academic Press: New York, 2015.
(3) Lyklema, J. Fundamentals of interface and colloid science: solid liquid interfaces; Academic Press: London, 1995.
(4) Bloomfield, V. A. DNA condensation by multivalent cations. Biopolymers 1997, 44, 269–282.
(5) Schoch, R. B.; Han, J.; Renaud, P. Transport phenomena in nanofluids. Rev. Mod. Phys. 2008, 80, 839–883.
(6) Bocquet, L.; Charlaix, E. Nanofluids, from bulk to interfaces. Chem. Soc. Rev. 2010, 39, 1073–1095.
(7) Werber, J. R.; Osuji, C. O.; Elimelech, M. Materials for next-generation desalination and water purification membranes. Nat. Rev. Mater. 2016, 1, 1–15.
(8) González, A.; Goikolea, E.; Barren, J. A.; Mysyk, R. Review on supercapacitors: technologies and materials. Renewable Sustainable Energy Rev. 2016, 58, 1189–1206.
(9) Park, M.; Ryu, J.; Wang, W.; Cho, J. Material design and engineering of next-generation flow-battery technologies. Nat. Rev. Mater. 2017, 2, 1–18.
(10) Choi, N. S.; Chen, Z.; Freunberger, S. A.; Ji, X.; Sun, Y. K.; Amine, K.; Yushin, G.; Nazar, L. F.; Cho, J.; Bruce, P. G. Challenges facing lithium batteries and electric double-layer. Angew. Chem., Int. Ed. 2012, 51, 9994–10024.
(11) Vayenas, C. G.; Brosda, S.; Pliangos, C. The double-layer approach to promotion, electrocatalysis, electrochemical promotion, and metal–support interactions. J. Catal. 2003, 216, 487–504.
(12) Grosberg, A. Y.; Nguyen, T. T.; Shklovskii, B. I. Colloquium: the physics of charge inversion in chemical and biological systems. Rev. Mod. Phys. 2002, 74, 329–345.
(13) Kilic, M. S.; Bazant, M. Z.; Ajdari, A. Steric effects in the dynamics of electrolytes at large applied voltages. I. Double-layer charging. Phys. Rev. E 2007, 75, 021502.
(14) Bazant, M. Z.; Storey, B. D.; Kornyshev, A. A. Double layer in ionic liquids: Overscreening versus crowding. Phys. Rev. Lett. 2011, 106, 046102.
(15) Kornyshev, A. A.; Qiao, R. Three-dimensional double layers. J. Phys. Chem. C 2014, 118 (32), 18285–18290.
(16) Šamaj, L.; Trizac, E. Wigner-crystal formulation of strong-coupling theory for counterions near planar charged interfaces. Phys. Rev. E 2011, 84, 041401.
(17) Fedorov, M. V.; Kornyshev, A. A. Ionic liquids at electrified interfaces. Chem. Rev. 2014, 114, 2978–3036.
(18) Righini, R. Ultrafast Optical Kerr-Effect in Liquids and Solids. Science 1993, 262, 1386–1390.
(19) Miranda, P. B.; Shen, Y. R. Liquid interfaces: A study by sum-frequency vibrational spectroscopy. J. Phys. Chem. B 1999, 103, 3292.
(20) Bouzigues, C. I.; Tabeling, P.; Bocquet, L. Nanofluids in the Debye layer at hydrophilic and hydrophobic surfaces. Phys. Rev. Lett. 2008, 101, 114503.
(21) Andanson, J. M.; Baiker, A. Exploring catalytic solid/liquid interfaces by in situ attenuated total reflection infrared spectroscopy. Chem. Soc. Rev. 2010, 39, 4571–4584.
(45) Biesheuvel, P. M.; Van Soestbergen, M. Counterion volume effects in mixed electrical double layers. *J. Colloid Interface Sci.* 2007, 316, 490–499.

(46) Mezger, M.; Schroder, H.; Reichert, H.; Schramm, S.; Okasinski, J. S.; Schoder, S.; Honkimaki, V.; Deutsch, M.; Ocko, B. M.; Ralston, J.; Rohwerder, M.; Stratmann, M.; Dosch, H. Molecular layering of fluorinated ionic liquids at a charged sapphire (0001) surface. *Science* 2008, 322, 424–428.

(47) Misra, A.; Kozma, K.; Streb, C.; Nyman, M. Beyond charge balance: counter-cations in polyoxometalate chemistry. *Angew. Chem., Int. Ed.* 2020, 59, 596–612.

(48) Zhu, Z.; Tain, R.; Rhodes, C. A study of the decomposition behaviour of 12-tungstophosphate heteropolyacid in solution. *Can. J. Chem.* 2003, 81, 1044–1050.

(49) Lopez, X.; Nieto-Draghi, C.; Bo, C.; Avalos, J. B.; Poblet, J. M. Polyoxometalates in solution: molecular dynamics simulations on the $\alpha$-$\text{PW}_{12}\text{O}_{40}^{3-}$ Keggin anion in aqueous media. *J. Phys. Chem. A* 2005, 109, 1216–1222.

(50) Dubois, L. H.; Nuzzo, R. G. Synthesis, structure, and properties of model organic surfaces. *Annu. Rev. Phys. Chem.* 1992, 43, 437–463.

(51) Misra, R. P.; de Souza, J. P.; Blankschtein, D.; Bazant, M. Z. Theory of Surface Forces in Multivalent Electrolytes. *Langmuir* 2019, 35, 11550–11565.

(52) Nesor, S.; Bechinger, C.; Leiderer, P.; Palberg, T. Finite-size effects on the closest packing of hard spheres. *Phys. Rev. Lett.* 1997, 79, 2348.

(53) Teng, L.; Tu, P. S.; Lin, I. Microscopic observation of confinement-induced layering and slow dynamics of dusty-plasma liquids in narrow channels. *Phys. Rev. Lett.* 2003, 90, 245004.

(54) de Jonge, N. Theory of the spatial resolution of (scanning) transmission electron microscopy in liquid water or ice layers. *Ultramicroscopy* 2018, 187, 113–125.

(55) Tao, X.; Hite, R. K.; MacKinnon, R. Cryo-EM structure of the open high-conductance Ca$^{2+}$-activated K$^+$ channel. *Nature* 2017, 541, 46–51.

(56) Liu, S.; Chang, S.; Han, B.; Xu, L.; Zhang, M.; Zhao, C.; Yang, W.; Wang, F.; Li, J.; Delpire, E.; Ye, S.; Bai, X.-c.; Guo, J. Cryo-EM structures of the human cation-chloride cotransporter KCC1. *Science* 2019, 366, 505–508.