Sustained CO$_2$-photoreduction activity and high selectivity over Mn, C-codoped ZnO core-triple shell hollow spheres

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Solar conversion of CO$_2$ into energy-rich products is one of the sustainable solutions to lessen the global energy shortage and environmental crisis. Pitifully, it is still challenging to attain reliable and affordable CO$_2$ conversion. Herein, we demonstrate a facile one-pot approach to design core-triple shell Mn, C-codoped ZnO hollow spheres as efficient photocatalysts for CO$_2$ reduction. The Mn ions, with switchable valence states, function as “ionized cocatalyst” to promote the CO$_2$ adsorption and light harvesting of the system. Besides, they can capture photogenerated electrons from the conduction band of ZnO and provide the electrons for CO$_2$ reduction. This process is continuous due to the switchable valence states of Mn ions. Benefiting from such unique features, the prepared photocatalysts demonstrated fairly good CO$_2$ conversion performance. This work is endeavoured to shed light on the role of ionized cocatalyst towards sustainable energy production.
Worldwide consumption of fossil fuels is unambiguously accountable for two critical issues, i.e. global warming by the augmented greenhouse gases (especially CO₂) emission and depletion of such non-renewable sources¹–⁴. Nowadays, it becomes a global appeal to develop sustainable energy grids based on renewable energy sources such as solar, wind, and geothermal energies⁵–⁸. Reliable conversion of CO₂ by the aid of solar light to energy-rich fuels (i.e. artificial photosynthesis) is an ideal solution for simultaneously relieving both energy and environmental crises⁹–¹¹. Albeit great efforts have been devoted to exploring the potential of artificial photosynthetic materials for CO₂ conversion, its efficiency is still unsatisfactory due to difficulties in activating thermodynamically stable CO₂¹²–¹⁴. Therefore, it is critical to construct suitable photocatalysts and coupled with active cocatalysts. Cocatalysts can effectively promote the separation and transfer of photogenerated charge carriers to attain appealing CO₂ reduction ability¹¹,¹３–¹⁶.

Transition metal-based materials, such as ions, oxides, sulfides, and phosphides, exhibited outstanding potentials as cocatalysts for both oxidation and reduction reactions¹⁷–²¹. They can not only effectively lower the barrier for CO₂ activation but also promote the separation of photogenerated charge carriers²²,²³. To date, most cocatalysts are in solid state. We wonder whether ionized cocatalysts which are embedded in the lattice of photocatalyst can also serve as a cocatalyst. They can become active sites for the photocatalytic CO₂ reduction. Due to their multiple oxidation states, they can swap electrons with photocatalyst as well as CO₂. Besides, the introduction of ionized cocatalyst avoids the difficulty of loading a cocatalyst on the photocatalyst. And the difficulty in maintaining intimate contact no longer exists.

Intricate hollow structures with tailored compositions have surpassed their solid counterparts in widespread applications. However, the fabrication of such intricate structures is still an arduous challenge. Multishell hollow structures (MSHSs) are characterized by manifold features such as large specific surface area, low density, as well as reduced mass and charge diffusion lengths²⁶,²⁷. All of these endow MSHSs with potential in multiple applications, such as photocatalysis, energy storage and conversion, drug delivery and sensors²⁵–²⁷. Concerning CO₂ photoreduction (PR), MSHSs possess prodigious advantages over their solid counterparts. These advantages particularly spring from esthetic shell layers. For one thing, their large surface area promotes CO₂ adsorption and surface redox reactions on both sides of the shells. For another, they provide a reduced path length for charge carriers, favoring charge migration and separation²⁸,²⁹. All these merits make MSHSs ideal support materials for anchoring photocatalysts aiming at designing elaborate catalytic system for CO₂ PR. However, the fabrication of such MSHSs is still an arduous challenge.

Herein, we develop a facile one-pot coordination polymer (CP) strategy to fabricate Mn, C-codoped ZnO core–triple shell hollow spheres (Mn, C-ZnO CTHSs). The process involves two steps: (i) formulation of Zn, Mn CP via interaction of salicylate ligand with the corresponding metal ions in the presence of polyvinylpyrrolidone (PVP) and (ii) air calcination of the resultant Zn, Mn CP (Fig. 1). The obtained photocatalyst with fascinating core–multishell morphology and tailored composition affords adequate CO₂ reduction activity, i.e. two times higher than commercial ZnO. Such enhanced activity is ascribed to the improved light absorption, favorable transfer of mass and charges across the thin shell layers, and aided CO₂ adsorption and activation. Interestingly, Mn species formed in the system as substituent of Zn ions in the ZnO lattice play an indispensable role during CO₂ PR. They act as active centers for CO₂ adsorption and subsequent favored CO₂ reduction by suppressing hydrogen evolution. Most importantly, in situ X-ray photoelectron spectroscopy results reveal that Mn species can restore their primal oxidation state after CO₂ PR without external treatment or subsidiary reducing agents. Instead, Mn⁹⁺⁺ species can extract one photogenerated electron from the system to regenerate the original Mn⁹⁺ moieties. The switching between Mn⁹⁺⁺/Mn⁹⁺ is recognizable by virtue of the multiple oxidation states of Mn and affordability of photogenerated electrons. These results suggest that ionized Mn can act as a durable and efficient ionized cocatalyst toward practical CO₂ PR.

Results and discussions

Formulation and characterization of Mn, C-ZnO CTHSs. The core–triple shell hollow spheres are obtained via a facile one-pot solvothermal preparation of Mn, Zn CP spheres (Mn, Zn-CPSs) followed by controlled air-annealing as depicted in Fig. 1. The CPSs as precursors were firstly formulated by the coordination interaction between Zn- and Mn-acetate and salicylate ligands (Fig. 1a). The carboxylate groups of salicylic acid play a significant role that they can coordinate to metal ions, enabling a delicate control over the composition. Moreover, this method is universal to most of the transition metals because non-selectivity is witnessed for salicylic acid³⁰.

Fourier transform infrared spectroscopy (FTIR) spectra of the Mn, Zn-CPSs (Supplementary Fig. 1) evidenced the interaction between deprotonated carboxylate groups of salicylate ligands and metal ions. Such interaction is assured after distinctive intensity drop of the corresponding C=O peak centered at 165⁷ cm⁻¹ and a slight shift to 165⁴ cm⁻¹³⁰. The deprotonation of the carboxylate groups is very critical to boost such coordination interaction. As a strong conjugated base, acetate anions can deprotonate the carboxylate groups of salicylic acid and thus enable reliable coordination between salicylate ligand and metal cations. To confirm this assumption, we replaced Zn- and Mn-acetates by their sulfates, nitrates, or chlorides and followed the same preparation recipe. All of them fail to fabricate such spherical architecture. We attribute such failure to the inability of the corresponding anions to deprotonate carboxylate groups of salicylic acid. After formation of Mn, Zn-CPSs, the generation of hollow structures derives from the calcination process. In this step, the Mn, Zn-CPS precursor is subjected to air calcination at
550 °C, a step that culminates the unique CTSHS morphology by heterogeneous contraction process (Fig. 1b, c).

Transmittance electron microscopy (TEM) (Fig. 2) images show Mn, Zn-CPSs as high uniform solid spheres (Fig. 2a) with an average diameter of about 1.3 μm and smooth surface texture. After calcination for 3 h, unexpectedly, the solid Mn, Zn-CPSs were converted into unique concentric Mn, C-ZnO CTSHSs, as noticed from the high contrast between the shell edges and centered hollow regions (Fig. 2b). The CTSHSs retain the spherical shape of the CPS precursor but the diameter is dramatically shrunk to 650 nm due to the contraction upon calcination. The surface turns rough, pervaded with crystallites with invasive pores. The size of the crystallites was measured to be 25 nm from high-resolution TEM image. Its porous structure was confirmed by N₂ adsorption–desorption isotherm (Supplementary Fig. 2 and Supplementary Table 1). Additionally, uniform distribution of the constituent elements ca. Zn, O, Mn, and C are observed from energy dispersive X-ray spectroscopy (EDS) elemental mapping of the calcined sample (as taken for sample calcined for 1h, Fig. 2c). The elemental distribution is presented in Supplementary Fig. 3 and the actual compositional percentages of Mn and C in 2% Mn, C-ZnO sample are 1.8 and 8.3%, as derived from elemental analysis and inductively coupled plasma atomic emission spectroscopy measurements, respectively.

To elucidate the evolution mechanism of the CTSHS morphology, TEM and field emission scanning electron microscopy (FESEM) images of the intermediates at different calcination stages were recorded (Fig. 3 and Supplementary Fig. 4). Before calcination, the solid spheres with distinct boundary are observed (Fig. 3a). At the initial stages of the calcination process (after 0.5 h), a significant temperature gradient (ΔT) along the radial direction of the CPSs was generated where the outermost building units are subjected to abrupt heating. Meanwhile, the carbon entities vanish and ZnO nanoparticles were crystallized at the outer surface. Thus, a core–shell structure (the first ZnO shell conjoined to the entire core) was generated (Fig. 3b). Once forming, the rigid ZnO shell plays an indispensable role in preventing further contraction of the outer diameter, even if the entire core suffers from continuous shrinkage upon the disintegration of organic moieties by further annealing, known as heterogeneous contraction31–34.

With extended calcination time, both the core and the emerged new shell experienced two opposite forces. Namely, inward cohesion force (F₁) originated from the sharp contraction of the interior core upon continuous disintegration of the skeletal carbon and outward adhesion force (F₂) emanated from the cling of adjacent crystallites. The balance between both forces culminated in varied structures. When the CPSs were calcined for 1 h, the temperature gradient gets smaller. And F₁ exceeds F₂, which leads to separation of the entire core from the outer shell, forming a yolk-shell structure (Fig. 3c and Supplementary Fig. 4b). With calcination continued for 2 or 3 h, the entire yolk underwent similar processes described before, forming the unique second (core–double shell, Fig. 3d and Supplementary Fig. 4c) and the third shell (core–triple shell, Fig. 3e and Supplementary Fig. 4d). It is noteworthy that the disintegration of the carbon moieties on both exterior and interior surfaces induces the formation of observable pores within the shell layers (Supplementary Fig. 4e), which facilitates the mass transfer of the reactants/products35,36.

All the calcined samples together with commercial ZnO (for comparison) show diffraction peaks (Supplementary Fig. 5) that are well indexed to Wurtzite hexagonal ZnO (JCPDS PDF Card No. 36-1451), which are in line with the (101) lattice fringes of ZnO (Fig. 3f). Adequate purity and good crystallinity of the prepared ZnO CTSHS samples are reflected from the sharp peaks and absence of secondary peaks even with 5% Mn loading. Additionally, compared with commercial ZnO, the lattice of the prepared catalysts is mostly contracted after the doping process (see Supplementary Note 1 for more discussion). These results suggest the successful doping of Mn and/or C within the crystal lattice of ZnO and Mn species are implemented to substitute Zn ions in the lattice.

**CO₂ PR activity of resultant samples.** The photocatalytic performance of comm. ZnO, CZ, and MCZ-x (CZ represents C-ZnO CTSHS sample, MCZ-x represents Mn, C-ZnO CTSHS samples with different percentages of Mn (x = 1, 2, and 5)) was evaluated in an online closed gas-circulation system under simulated solar light with a Quartz and Pyrex glass hybrid reaction cell (Supplementary Fig. 6). Ambient conditions were kept, and neither photosensitizer, precious cocatalyst nor sacrificial reagents were used. Carbon
monoxide (CO) was the predominant reduction product, whereas oxygen (O$_2$) was the oxidation product (Fig. 4). The 2% Mn, C-ZnO CTSHS photocatalyst (MCZ-2 in Fig. 4a) afforded the highest PR activity with CO yield rate of 0.83 μmol g$^{-1}$. When Mn loading is too high, i.e. 5%, excessive defect states are introduced into the lattice of ZnO, which act as local recombination centers for photogenerated charge carriers$^{38}$. H$_2$ gas was hardly detectable, suggesting that the competitive H$_2$ evolution from water is inhibited without the help of noble metal cocatalyst. The initial detected O$_2$ gas came from the input high-purity CO$_2$. As the CO$_2$ PR proceeded, the amount of O$_2$ increased gradually, indicating that water was photooxidized to O$_2$ (Fig. 4b). Taking into account the sluggish kinetics of water oxidation and the need of four hole equivalents to produce one oxygen molecule$^{39}$, it is reasonable to observe a lower yield of O$_2$ (oxidation product) than CO (reduction product).

The apparent quantum efficiency (A QE) of CO is defined by the ratio of the effective electrons used for CO production to the total input photon flux$^{40,41}$. It was measured by applying a 300 W xenon arc lamp with monochromatic light band-pass filters (λ = 350, 365, 380, 400, 420, and 450 nm; Beijing Perfectlight, China). The measured light intensity per unit area (E), the CO production yield (M$_{\text{CO}}$), and the calculated CO production rate (R$_{\text{CO}}$). AQE at different monochromatic wavelengths over the 2% Mn, and C-ZnO sample are shown in Table 1. Further, the wavelength dependence of the AQE and the UV–vis absorption spectrum of 2% Mn, C-ZnO sample is shown in Fig. 4c. Clearly, the trends of AQE match well with the light absorption spectrum, indicating the photocatalytic feature of the CO$_2$ reduction reaction.

The recyclability and stability of MCZ-2 for photocatalytic CO$_2$ reduction were also investigated (Supplementary Fig. 7). After four cycles, the decay of CO production yield is hardly perceptible. Meanwhile, the recycled photocatalyst shows identical phase composition as the fresh sample as confirmed from X-ray diffraction (XRD) results (Supplementary Fig. 8). These findings unambiguously disclosed the promising potential of the prepared CTSHS photocatalysts for CO$_2$ PR.

Control experiments showed that no products were detected in the absence of photocatalysts and/or CO$_2$ either in dark or under light irradiation, suggesting that CO is exclusively generated from CO$_2$ PR. We performed isotopic labeling experiment using $^{13}$CO$_2$ as isotopic tracer to investigate the carbon source of the obtained products. As shown in Fig. 4d, the total ion chromatographic signal around 11.57 min corresponds to CO, which produces three signals in the mass spectra (Fig. 4e). The main MS signal at $m/z = 29$ belongs to $^{13}$CO and the others at $m/z = 13$ and $m/z = 16$ are assigned to the fragments of $^{13}$CO, i.e. $^{13}$C and $^{16}$O, respectively. This result confirms that CO is indeed derived from CO$_2$ reduction over MCZ-2 rather than any other carbon source$^{42,43}$. In addition, the total ion chromatographic signals around 2.67 and 3.55 min are assigned to O$_2$ and N$_2$, respectively (Supplementary Fig. 9).

**Photocatalytic mechanism for enhanced performance.** Appreciable enhancement of the visible light absorption was sustained by the fascinating hollow morphology, thanks to the multiple reflections of the incident light inside the interior cavities of the fascinating core–triple shell hollow structure$^{44}$ (inset of Fig. 5a). Moreover, the codoping process introduces impurity levels within the forbidden gap of ZnO, which indeed contributes to narrowing the bandgap (as depicted from Tauc plots, Supplementary Fig. 10) and enhancing the light-harvesting ability of the doped samples. A stepwise improvement of the light absorptivity of the prepared photocatalysts is observed in line with Mn-content (Fig. 5a). This finding should be correlated to the fact that more impurity states are developed within ZnO bandgap with increasing Mn doping. In addition, the doping with Mn could initiate other electronic transitions such as ligand to metal charge transfer, metal to ligand charge transfer, and d–d transition$^{45–47}$. All these transitions are accounted for the whole system light harvesting, which leads to enlarged absorption response$^{48,49}$.

Furthermore, the photoluminescence (PL) spectra of the prepared photocatalysts are step-wisely quenched after Mn and C doping. This reveals that suppressed charge recombination is achieved by doping. The 2% Mn, C-ZnO sample exhibits the lowest PL behavior (Fig. 5b), benefiting from two factors. One is (i) the defect state. The rest is (ii) the core–shell morphology and porous structure. The former could act as trapping centers, quenching the electron–hole inhalation process. Whereas the latter reduces both diffusion lengths of charges and reactants, thus minimizing charges recombination opportunities and providing accessible channels for mass/charge transfer$^{50}$. Time-resolved

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**Fig. 3 TEM images of Mn, Zn-CPSs. a** Before calcination and after calcination for **b** 0.5 h, **c** 1 h, **d** 2 h, and **e** 3 h. **f** Depicts the HRTEM of Mn, C-codoped ZnO with lattice fringes of 0.248 nm corresponds to the (101) plane of Wurtzite ZnO.
photoluminescence (TRPL) results also demonstrate that 2% Mn, C-ZnO can retard charge carriers recombination and prolong their lifetime (average charge carrier-lifetime ca. 1.1, 0.9, and 0.8 ns for 2% Mn, C-ZnO, and Comm. ZnO, respectively, Fig. 5c and Supplementary Table 3). These results indeed affirm the potential of Mn doping to delay charge recombination.

In addition, the prepared samples (C-ZnO and Mn, C-ZnO-CTSHSs) attain very low resistance for charges transfer ($R_{CT}$) process compared to comm. ZnO sample (as speculated from the electrochemical impedance spectroscopy measurements, Supplementary Fig. 11, where smaller arc diameter represents lesser resistance). Obviously, the fascinating architecture favors charge carrier separation and diffusion from bulk to surface region through few nanometer-thick shells. Especially, the Mn-doped sample shows the least $R_{CT}$, benefitting from charge trapping action exerted by the doping-based defect states (at optimum content) and the following interfacial charge transfer51,52.

Besides, Mn species could be beneficial for enhancing the adsorption of CO$_2$. To authenticate such a hypothesis, the CO$_2$ adsorption isotherms for C-ZnO and 2% Mn, C-ZnO-CTSHSs are provided in Fig. 5d. Obviously, the 2% Mn-doped sample shows much more adsorption affinity (merely twice) toward CO$_2$ than the undoped sample does. This result coincides well with the photocatalytic activity of both samples. The higher the CO$_2$ adsorption affinity of a photocatalyst is, the better the PR activity will be expected. Therefore, it is reasonable to notice such enhanced activity for Mn-doped photocatalyst. The improvement of CO$_2$ adsorption after Mn doping was also reported for Mn-In$_2$S$_3$ tested for CO$_2$ electroreduction53.

### Table 1: The measured $E$, $M_{CO}$, and the calculated $R_{CO}$, AQE at different monochromatic wavelengths over the 2% Mn, C-ZnO sample.

| $\lambda$ (nm) | 350 | 365 | 380 | 400 | 420 | 450 |
|---------------|-----|-----|-----|-----|-----|-----|
| $E$ (mW cm$^{-2}$) | 5 | 7 | 8 | 7 | 14 | 12 |
| $M_{CO}$ (ppm) | 48 | 75 | 48 | 42 | 33 | 25 |
| $R_{CO}$ (μmol h$^{-1}$) | 0.033 | 0.051 | 0.033 | 0.029 | 0.022 | 0.017 |
| AQE (%) | 0.028 | 0.030 | 0.016 | 0.015 | 0.0056 | 0.0047 |
Two plausible factors contributing to the high CO₂ adsorption affinity are proposed. One is the higher content of surface oxygen-species (see XPS interpretation, Supplementary Note 2 for details) than C-ZnO sample. The other is the potential of Mn species replaced Zn ions in the lattice to activate CO₂ molecules.

To further explore the functional role of Mn, in situ irradiated XPS (ISI-XPS) analysis for Mn, C-ZnO CTSHS sample saturated with CO₂ was conducted under both dark and irradiated conditions. The high-resolution XPS peaks are demonstrated in Fig. 6 and Supplementary Fig. 12 together with the same peaks from bare sample (without CO₂ treatment). The XPS spectra further indicate the successful doping of Mn within the crystal lattice of ZnO, originating from the distinctive Mn 2p and Mn 3s peaks appeared in the XPS profiles and the shift of Zn 2p½/2 peak (see Fig. 6 and Supplementary Fig. 12 for more details about the chemical state of Mn within the ZnO lattice). The presence of Mn is believed to promote catalytic conversions owing to their ability to share electrons with reactant molecules and/or intermediates. Besides, doping gives rise to defect states and impurity levels. They not only reduce the bandgap of the semiconductor but also improve the light absorptivity of the photocatalyst. The defect states could support the charge separation process through obstructing the electron–hole recombination by electron trapping. Specifically, introduced transition metal ions could act as distinctive active sites or enrich present active sites, which evidently support reactants’ adsorption, activation, and further chemical conversion.

The loading of CO₂ molecules onto the surface was assured by emergence of a tiny M-CO₃ peak in C 1s XPS (Supplementary Fig. 12d), suggesting the successful bonding of CO₂ molecules to metallic sites. To probe how Mn ions could assist activation of CO₂ molecules during the PR process, we calculated the average oxidation states (AOS) of Mn under test according to the following equation:

\[ \text{AOS} = 8.95 - 1.13 \Delta S \]  

where \( \Delta S \) is the multiple splitting of Mn 3s peak (Fig. 6a, b). By substituting \( \Delta S \) values in Eq. (1), we get a unit increase in the AOS for sample saturated with CO₂ and kept in darkness. Then, a decrease in AOS by half a unit is observed for sample saturated with CO₂ and irradiated with light. The significant increase in AOS of Mn after CO₂ loading and before illumination could be explained in terms of electron transfer from Mn centers to adsorbed CO₂ molecules. Our system, we noticed that the primal AOS of Mn in CO₂-loaded photocatalyst can be restored again under light irradiation. This is due to photogenerated electrons transfer from ZnO-conduction band (CB) to Mn\(^{n+1}\)+ centers to regenerate the primal Mn\(^{n+1}\) again (Fig. 6c). This light-switchable conversion between Mn\(^{n+1}\)/Mn\(^{n+1}\)+ is very critical for restoring the original activity of the Mn-doped catalyst without external treatment or activation steps. In addition, this conversion holds a promising potential for all-in-one CO₂ reduction technology. Concisely, Mn\(^{n+1}\)+ species can capture the photogenerated electrons from the CB of ZnO, yielding Mn\(^{n+2}\), which functions as the active centers for CO₂ adsorption and activation via one electron-transfer process to yield CO\(_2\)\(^-\) and Mn\(^{n+1}\)+. At the same time, if more electrons are captured, the generation of CO becomes possible. This finding authenticates the role of Mn species as ionized cocatalyst in the system. Because we measure the change of AOS, further investigations are required to elucidate the actual functional oxidation state involved in CO₂ activation.

In fact, the regeneration of the Mn species after the activation step is very critical to maintain the activity. Moreover, the charge trapping action is of great essence for charge carrier’s separation.
Thus the optimal doping from Mn results in lowest charges recombination as revealed from PL results. In this vein, Mn functions as Lewis base center, which supports the adsorption of acidic CO₂ molecules and further transfer of one electron to culminate activated CO₂· (Fig. 6c). Such a step is followed by proton-coupled electron-transfer process to produce the desired molecules (Supplementary Eqs. (1) – (7)). The adsorption, activation and subsequent reduction of CO₂ molecules over Mn-doped ZnO photocatalyst are further confirmed by in situ diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS) results (Supplementary Fig. 13 and Supplementary Note 3), which assure the adsorption, activation, and successful conversion of activated CO₂ molecules onto CO at the photocatalyst surface (Fig. 6c).

All these results unambiguously disclosed the potential of the prepared and light-switchable ionized cocatalyst (Mnⁿ⁺/Mnⁿ⁺+1) integrated C-ZnO CTSHSs as an efficient photocatalyst for CO₂ PR under ambient conditions without sacrificial reagents, precious cocatalysts, or external treatment set up for activity restoration.

In summary, core–triple shell Mn, C-codoped ZnO hollow spheres were rationally fabricated via a one-pot synthetic protocol. In addition, they were tested as a highly active and selective catalyst for CO₂ PR under simulated sunlight and ambient conditions. The unique core–triple shell hollow structures codoped with Mn and C hold manifold structural and functional features. These unique features culminated improved light absorption capability, enhanced charge separation and migration efficiency, suppressed e⁻/h⁺ pairs undesired recombination probability and abundant adsorption and activation active sites for effective photoredox catalysis. All of these accounted for the superior activity of the prepared photocatalyst. The Mn species played a significant role to activate CO₂ molecules through an electron-transfer mechanism. We have observed that the ionized Mn species can restore their primal oxidation state and hence their activity by virtue of the photogenerated electrons at the ZnO CB. Therefore, it can be operated as a light-switchable ionized cocatalytic system. These findings will support the design of efficient photocatalytic materials for sustainable clean energy production and beyond. The present study sheds light on the significance of hollow structure materials as an active photocatalyst for CO₂ reduction as well as the importance of light-switchable ionized cocatalysts during this process.

Methods
Materials. All chemicals were of analytical grade and used without additional purification. Zinc acetate dihydrate (Zn(CH₃COO)₂·2H₂O) and absolute ethanol (EtOH) were purchased from Beijing Sinopham Chemical Reagent Co., Ltd.

Fig. 6 Interpretation of the CO₂ reduction mechanism over Mn, C-ZnO photocatalyst. In situ XPS analysis showed Mn 3s profiles for a untreated and b CO₂-adsorbed Mn, C-ZnO CTSHS samples. The multiple splitting of Mn 3s (ΔS) is used to calculate average oxidation state of Mn ions within the structure (Eq. (1)). c Schematic illustration of CO₂ activation and reduction over Mn, C-codoped ZnO sample. The CO₂ molecules are firstly activated at the confined Mn active centers to the highly active radical anion. Such intermediate anion is then subjected to successive electron-coupled proton transfer process (electrons from ZnO CB and protons from water) and finally produces CO. The photogenerated holes are consumed for water oxidation (oxygen evolution reaction).
Manganese acetate tetrahydrate (Mn(CH₃COO)₂·4H₂O) and salicylic acid (C₆H₅COOH) were purchased from Tianjin BODI Chemical Co., Ltd. PVP (C₆H₈O₇·nH₂O; MW~20,000) was supplied by Aladdin Industrial Corporation, Shanghai, China.

**Synthesis of Mn, C-ZnO CTSHSs.** The Mn, Zn-CPs were firstly prepared by a solvothermal method. Typically, 1 mmol of Mn(CH₃COO)₂·2H₂O, x mmol of Mn(CH₂COO)₂·4H₂O corresponding to the desired Mn²⁺/Zn²⁺ molar ratio (0, 0.01, 0.02 or 0.05), and (2 + x) mmol of salicylic acid were magnetically dissolved in 50 mL of EtOH. Afterwards, a calculated amount of PVP (to keep the concentration ratio of MnPVP/MoPVP = 5, M of PVP was calculated based on the molecular weight of its monomer) was added to the above solution and further stirred for 30 min. The mixture was transferred into a 100 mL of Teflon-lined stainless steel autoclave, heated at 160 °C for 12 h. Then air-cooled to room temperature. The precipitate was separated by centrifugation, washed three times with EtOH, and dried under vacuum at 60 °C overnight. The as-obtained Mn, Zn-CPs were calcined at 550 °C for 3 h by 5 °C/min under air atmosphere. The samples with 0, 1, 2, and 5% Mn were denoted as C-ZnO (CZ), 1% Mn, C-ZnO (MCZ-1), 2% Mn, C-ZnO (MCZ-2), and 5% Mn, C-ZnO (MCZ-5), respectively.

**Characterization.** TEM, HRTEM, and elemental mapping were obtained by Titan G2 60–300 operated at 300 kV. FESEM images were taken by JSM-7500F, JEOL, Japan, equipped with X-Max 50 EDX unit (Oxford Instruments, UK). XRD patterns of the prepared samples were recorded using Cu Kα radiation with a monochromatic wavelength of 1.5406 Å at a scan rate of 0.05 °s⁻¹. The surface composition and chemical states of C-ZnO and 2% Mn, C-ZnO CTSHS samples were analyzed by XPS analysis (Thermo ESCALAB 250). IAS-XPS analysis for 2% Mn, C-ZnO CTSHS sample saturated with CO₂ was conducted in darkness as well as under light irradiation to probe the oxidation state change of Mn during CO₂ PR. The 2% Mn, C-ZnO CTSHS sample was firstly degassed at 150 °C for 4 h to remove pre-adsorbed species. Afterwards, it was saturated with CO₂ by continuously purging with pure CO₂ for 4 h prior to the XPS test. A 3 W 365 nm LED was used as the light source during IAS-XPS test. The UV–visible light absorption spectra of the samples were recorded by UV-2600, Shimadzu spectrophotometer (Japan) where BaSO₄ was used as the reference. The absorbance spectra of the samples deposited on FTO glass, and the three electrodes were immersed in 0.5 M Na₂SO₄ aqueous solution as electrolyte. Pt wire and Ag/AgCl (saturated KCl) utilized as counter and reference electrodes, respectively.

**Photocatalytic CO₂ reduction and isotope-labeling measurement.** The photochemical reaction, 1 mL of the mixed gas was taken out from the reactor and analyzed by gas chromatography-mass spectrometry (TRACE 1300 and ISQ 7000, Thermo scientific, USA) equipped with the column for detecting the product of 13CO₂ (TG-BOND Siv 5 A, 30 m × 0.32 mm × 30 µm, Thermo Scientific, USA). The column was maintained at 25 °C for 15 min and then heated to 180 °C at 30 °C min⁻¹ and maintained for 1 min. The temperatures of the injector, ion source, and MS transfer line were set to 200, 230, and 250 °C, respectively.

**Data availability**

The raw data of Figs. 4, 5, 6a, b, Supplementary Figs. 1, 2, 5, 7 and Tables 1 and 3 generated in this study have been deposited in the figshare database [https://doi.org/10.6084/m9. figshare.14094753]. Source data are provided with this paper.

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Acknowledgements
This work was supported by NSFC (51872220, 5196135303, 51932007, 21871127, 52073233, U1905215, and 52030312) and the Fundamental Research Funds for the Central Universities (WUT: 2020I027).

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M.S., F.X., P.K., I.L., L.Z. and J.Y. conceived and designed the experiments. M.S., F.X., P.K., J.L., S.W., L.Z., and J.Y. conducted data characterizations of the materials. M.S., F.X., I.L., S.W., L.Z., and J.Y. revised the manuscript. All authors discussed the results and commented on the manuscript.

Competing interests
The authors declare no competing interests.
Additional information
Supplementary information The online version contains supplementary material available at https://doi.org/10.1038/s41467-021-25007-6.

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Peer review information Nature Communications thanks Jeffrey Chi-Sheng Wu and the other, anonymous, reviewer(s) for their contribution to the peer review of this work.

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