Electrical and magnetic response of a phosphate glass - NiFe$_2$O$_4$ composite. A novel magnetic sensor design

Abstract

NiFe$_2$O$_4$ spinel – phosphate (Bi-Ba-Li) glass composite was synthesized by solid-state reaction. New crystalline magnetic phases was developed inside the glassy matrix through a controlled heat treatment. New NF-LBPB$_4$ magnetic nanocomposite material was study. Complex impedance analysis has shown that mobile ions inside the matrix induce the development of complementary crystalline phases resulting in a composite material with excellent magnetic response. An innovative homemade device was designed to test the nanocomposite magnetic response.

Introduction

Nowadays, magnetic nanoparticles materials increase their interest because they lead to new technological applications. A step forward is magnetic-glassy nanocomposite materials because they give the opportunity to obtain devices with very different shapes retaining their electrical and magnetic behaviour. Additionally, glassy matrix acts as a protection of the nanocrystalline magnetic material. Liquid phase synthesis remains inside the glassy matrix through a controlled heat treatment resulting in a composite material with excellent magnetic response. An innovative homemade device was designed to test the nanocomposite magnetic response.

Material synthesis

NiFe$_2$O$_4$ [NF] synthesis

In this work NF (NiFe$_2$O$_4$) pure powder was prepared by solid-state reaction method. NiCO$_3$ (II) (Riedel-de Haen AG. Seelze-Hannover) and Fe$_2$O$_3$ (Sigma-Aldrich ≥99%) powder were use to synthesis NF. Starting from the nominal composition every component was weighted and mixed in an agate mortar. The powders were well mixed and placed in a platinum crucible and decarboxylated at 800°C for 2h in an air atmosphere and followed by 2h at 1100°C. The obtained powder was pressed in pellets of 10 mm diameter and 2mm thickness at 10.2 metric ton/cm$^2$ and sintered at 700°C for 2h. Finally, the pellets were grind again.

Phosphate glasses synthesis

Ternary glasses of composition: 19BaO (74P$_2$O$_5$, 7Bi$_2$O$_3$) and 42Li$_2$O 18BaO (36P$_2$O$_5$, 4Bi$_2$O$_3$) (mol%) was prepared by a standard melt quenching technique. From now [LBPB] and [LBPB4].

Composites NF-LBPB synthesis

The prepared glasses were ball-milled and the resulted powder was sieved with a 150μm mesh sieve to obtain the small fraction of particles. The glassy powder was mixed with NF powder in a ratio of: 89%wt/11%wt. The mix was heated at 700°C for 30min.

Composite NF-LBPB4 synthesis

Powdered [LBPB4] glass was mixed with a [NF] in a ratio of 89%wt to 11%wt respectively. Both powders were mixed in an agate mortar a new glassy material was prepared by the standard melt quenching technique. First, it was heated for 1 h a 1353K, shaken frequently to ensure homogenization. The melt was poured on an aluminum plate at 0°C and the obtained solid drops were held for two hours a 473K to remove the mechanical stress and 3hrs at 703K to allow the grows of [NF] crystallites inside the glassy matrix.
Materials characterization and results

Structural

The glassy nature of the resulting solids by quenching technique was tested using X-ray diffraction (XRD) analysis and differential temperature analysis (DTA). DTA curves were recorded using a Rigaku Thermoflex TG 8110 associated with Thermal Analysis Station TAS 100, the T_g (glass transition temperature) of each sample was determined with a heating rate of 10 K min⁻¹ in the range 25°C to 400°C using 20–30 mg of glass samples finely grounded in an agate mortar.²³ Figure 1 shows the X-ray diffraction patterns that confirms the single phase material. All the diffraction peaks can be indexed to the cubic spinel structure of JCPDS card no. 74-2081 with the presence of (220), (311), (400), (422), (511), (440), and (622) planes, with small proportion of impurities. The lattice parameter was in good agreement with standard value. Figure 2 & Figure 3 show the DTA scan and the corresponding T_g for each glassy sample. From the results in these figures we learn that the T_g value decreases 10% when lithium oxide is incorporated in the phosphate glassy matrix. Also, Figure 3 shows two extra peaks after its T_g which belong to the devitrification phenomenon.

![X-ray diffraction patterns](image)

Figure 1 [NF] X-ray diffraction pattern (pink); NF/LBPB composite (blue) X-ray diffraction pattern; [LBPB4] (red) and [LBPB] (black) X-ray diffraction pattern.

Electrical properties

Every glassy sample was polished with a sand paper to obtain disks with two parallel mirror surfaces with thickness ranging between 0.5 to 0.7mm and coated with a thin layer of silver paint to having a proper electrical contact. Impedance measurements were carried (Agilent LCR meter 4284A; 20Hz–1MHz; AC voltage amplitude signal of 0.80 V) in a temperature range between room temperature and (T_g−15)°C. Electrical measurements was done registering the magnitude of the sample impedance |Z| and the phase angle φ, using the impedance meter. Show the complex permittivity ε'(ω) and the dielectric loss (tan δ) measured as a function of the temperature for the pristine composite NF-LBPB4 and NF/LBPB4 treated 1h to 703K. The values of the complex permittivity are calculated using the following equations:

\[
\varepsilon' = \frac{L \cdot \tan \theta \cdot \tan^2 \omega_{100}}{A \cdot Z \cdot [f^2 + \pi^2 f^2 \cdot \varepsilon_0]^{100}}
\]

\[
\varepsilon'' = \frac{L \cdot \tan \theta \cdot \tan^2 \omega_{100}}{A \cdot Z \cdot [f^2 + \pi^2 f^2 \cdot \varepsilon_0]^{100}}
\]

Figure 4 shows that the ε’ is lower when the material is heat treated and also the dielectric loss which decrease more confirming lower energy losses than in the pristine sample. The Maxwell-Wagner model for the heterogeneous structure explain σ the ε’ dispersion phenomenon observed at low frequency. We assume that the grain boundary present in the composite are smoothed after the heat treatment due to a cation interchange between the combined materials. Additionally, Figure 5A & Figure 5B show that after the heat treatment crystallites are larger enough to be revealed by a second resistance response clearly observed in Figure 5B. However, as we mentioned above the interface between both materials is less...
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The glassy matrix is trapped around the crystalline phase given new crystalline structures as X-ray diffraction pattern shows in Figure 1.

**Figure 4** Real part of the complex permittivity of: A) NF-LBPB4 pristine composite and B) NF-LBPB4 treated 1 h at 703K.

**Figure 5** Impedance imaginary part: A) NF-LBPB4 pristine composite and B) NF-composite heat treated.

**Figure 6** Conductivity spectra: A) NF-LBPB4 pristine composite and B) NF-LBPB4 heat treated.

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Magnetic properties

Figure 7 shows the magnetization loops (T-H) measured at room temperature.\textsuperscript{20} It exhibits typical ferromagnetic behavior and it reaches a saturation magnetization values $3M_s$ (pristine composite; glass ceramic)\textsuperscript{-} $M_s$ (heat treated composite, after 1 or 2 hours). The composite prepared as a glass-ceramic has the same response as the pristine composite. In order to obtain such response we have design a homemade magnetic analyzer. A sensor with a fine adjustment screw type system was built like describe the following Figure 8(A-D), using the same circuit, the measurement is made with air core and winding of 100/100 turns.\textsuperscript{21,22} The magnetization curve of each material is determined based on which the flow is determined by the supply voltage. The circuit used is shown in Figure 8:

$$V_i = 4.44 \times f \times N \times \Phi \max \Rightarrow \Phi \max = \frac{V_i}{4.44 \times f \times N}$$  \hspace{1cm} (4)

$$H = \frac{N \times P \times I_s}{i_j}$$  \hspace{1cm} (5)

Figure 7 Magnetic hysteresis loops at room temperature of NF-LBPB4 composites pristine and heat treated at different times.

![Figure 7 Magnetic hysteresis loops at room temperature of NF-LBPB4 composites pristine and heat treated at different times.](image)

The current is measured on the primary winding and the voltage on the secondary winding, thus achieving that both values are not affected by the insertion of the instruments. The current determines the magneto motive force and the voltage the conversion flow. A device was designed to check the operational material behavior using 0.27 mm thick oriented grain iron as reference. The system has the

Figure 8 A) Sensor design; B) Sensor built; C) sample location; D) oscilloscope ScopeMeter Fluke199) 200 MHz 2.5 GS / s, double input.

![Figure 8 A) Sensor design; B) Sensor built; C) sample location; D) oscilloscope ScopeMeter Fluke199) 200 MHz 2.5 GS / s, double input.](image)

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following components:

A. Drag system of neodymium magnets.

B. Device where the samples are placed and voltage taken.

C. The programmable control automaton (Figure 9).

Figure 9 Novel devices for comparison magnetic response.

The device has a 24 volt direct current motor, rack and chain system that convert the rotary movement into linear. The system has a permanent magnet and limits of career and position sensors for the movement and measurement.\textsuperscript{23,24} Also, the system has a second element where the samples are placed and where is the voltage induced. The voltage generated by the samples is compared to the passage of the permanent magnets. We use first silicon steel whose characteristics are well known and next our glassy composite materials. It is important to emphasize that both shape and dimensions of samples are thoroughly determined. The generated voltage on the driver depends on the magnetic field intensity due to the speed of the magnets and the magnetic circuit that surrounds the driver. As an example, Figure 10 shows the result of one measure.

First signal of $7.20 \times 10^{-3}$ s – Intensity: from 2mV to 50mV

Second signal of $8.40 \times 10^{-3}$ s – Intensity: from 2mV a 194mV

We took 9727 measurements during 3843 seconds in order to obtain an average response.
Discussion

The growth of the crystalline phase inside the glassy matrix improves the magnetic response of the composite material. The resulting material’s $T_c$ is high enough to permit the proposed technological use. From the electrical results we show that a smooth inter phase between the magnetic crystallites and the glassy matrix is obtained after an appropriated heat treatment assisted by the cation interchange. The crystallites obtained are larger enough to be observed and the mobile ions in the glassy matrix trapped around the crystalline phase give a new crystalline structures. As a result, the composite material developed has the advantages of the glassy materials with the magnetic properties necessary to be applied in green electrical generation for example (as part of a stator). The innovative homemade device designed allows the magnetic characterization of our composite material. Considering that it was built with standard laboratory equipment, it is an excellent opportunity to test the material performance in comparison to very well known magnetic materials as for example silicon steel that we used.

Conclusion

In this work we synthesized the NiFe$_2$O$_4$ spinel by solid-state reaction and the phosphate (Bi-Ba-Li) glass by melt quenching technique. Applying a mechanical mix and several heat treatments an innovative nanocomposite was developed. Additionally, the heat treatments led to growth new crystalline magnetic phases inside the glassy matrix. The NF-LBPB4 magnetic nanocomposite material was thoroughly study. Complex impedance analysis allows to evidence that are the cations inside the matrix which induce the development of complementary crystalline phases. The composite material obtained has an excellent magnetic response. In order to evaluate the composite magnetic behavior an innovative homemade device was designed. We present technical details to build it.

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Conflicts of interest

Author declares that there is no conflict of interest.

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