Anharmonic phonon spectra of PbTe and SnTe in the self-consistent harmonic approximation.

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At room temperature, PbTe and SnTe are efficient thermoelectrics with a cubic structure. At low temperature, SnTe undergoes a ferroelectric transition with a critical temperature strongly dependent on the hole concentration, while PbTe is an incipient ferroelectric. By using the stochastic self-consistent harmonic approximation, we investigate the anharmonic phonon spectra and the occurrence of a ferroelectric transition in both systems. We find that vibrational spectra strongly depend on the approximation used for the exchange-correlation kernel in density functional theory. If gradient corrections and the theoretical volume are employed, then the calculation of the free energy Hessian leads to phonon spectra in good agreement with experimental data for both systems. In PbTe, we reproduce the transverse optical mode phonon satellite detected in inelastic neutron scattering and the crossing between the transverse optical and the longitudinal acoustic modes along the ΓX direction. In the case of SnTe, we describe the occurrence of a ferroelectric transition from the high temperature Fm3m structure to the low temperature R3m one.

I. INTRODUCTION

Thermoelectric materials are appealing for their capability of converting heat into electric power, and vice versa. Used in conjunction with clean sources of energy, such as solar radiation, such devices may be an alternative solution for the increasing global energy demand, and other global issues such as global warming.

The development of efficient thermoelectric devices is crucial to understand the lattice-dynamics and thermoelectric efficiency of materials. From a theoretical point of view, several approximations have been developed in the last years to tackle this problem and it is finally becoming possible to describe anharmonic effects beyond the perturbative regime.

Among thermoelectrics, PbTe and SnTe have drawn attention due to some of their interesting properties. Both systems have high figures of merit, turning them into effective thermoelectrics. Furthermore, at room temperature they have simple NaCl-like structures. The simplicity of their structure and the importance of anharmonic effects to describe their lattice-dynamical properties make them an ideal playground to validate non-perturbative theoretical approaches to the anharmonic problem.

Beside its thermoelectric properties, SnTe displays an intriguing ferroelectric transition at low temperatures. This transition towards a rhombohedral structure occurs when the transverse optical (TO) modes at the Brillouin zone (BZ) center softens during cooling. Past and recent experiments measured different transition temperatures, ranging from 0 K to values around 120 K. Such a variation in Te is due to the change in the intrinsic doping. On the other hand, PbTe does not undergo a phase transition at low temperature, although it has an incipient ferroelectric nature. Moreover, recent INS experiments reported that PbTe exhibits a phonon satellite peak close to zone center, a clear fingerprint of strong anharmonicity.

From the theoretical point of view, both systems have

\[ ZT = \frac{S\sigma T}{k} \]

\[ S \text{ is the Seebeck coefficient, } T \text{ the temperature, and } \sigma \text{ and } k \text{ are the electronic and thermal conductivities, respectively. The higher is the figure of merit, higher is the efficiency of a thermoelectric device.} \]

In these systems anharmonic effects play an important role as they decrease the thermal conductivity via phonon-phonon scattering and, consequently, increase ZT. In addition, some thermoelectric compounds undergo displacive second order phase transitions driven by soft modes. Close to the second order phase transition the phonon frequencies become very soft and the potential felt by the ions strongly anharmonic. Hence, a detailed non-perturbative treatment of anharmonicity is crucial to understand the lattice-dynamics and thermoelectric efficiency of materials.
been studied in the past by using ab-initio calculations, as well as molecular dynamics based methods\textsuperscript{17,19–22}. The majority of them use an non-perturbative approach since previous calculations based on perturbation theory report structural instabilities that are not present in the measurement\textsuperscript{6,19}. Some of those methods methods give good agreement with the experiment, specially for PbTe. However, the results may be affected by a conjunction of factors, such as the volume and the exchange correlation functional used in the calculations\textsuperscript{23}. Moreover these methods do not include the quantum nature of the ions that could be relevant at low temperature.

In this work we present the anharmonic phonon spectra of PbTe and SnTe as function of temperature using the stochastic self-consistent harmonic approximation (SSCHA)\textsuperscript{8,9,24}. We apply the method to both systems and determine the magnitude of anharmonic effects on vibrational spectra and on the ferroelectric transition.

This paper is presented in the following order: First, in section II we introduce the theoretical background and methodology applied in our calculations. Then we introduce the stochastic self-consistent harmonic approximation (SSCHA)\textsuperscript{8,9,24} that includes both thermal and quantum fluctuations of the ions, and, in this framework, the evaluation of the free energy Hessian\textsuperscript{24}. Section III presents the parameters used in our ab-initio calculations. The main results concerning the harmonic and anharmonic phonon dispersions, and comparison between our calculations and experimental data are described in section IV.

II. THEORY

We study the lattice dynamic of PbTe and SnTe in the Born-Oppenheimer (BO) approximation, thus we consider the quantum hamiltonian for the atoms defined by BO potential energy \( V(R) \). With \( R \) we are denoting in component-free notation the quantity \( R^{a_s}(l) \), which is a collective coordinate that completely specifies the atomic configuration of the crystal. The index \( a \) denotes the Cartesian direction, \( s \) labels the atom within the unit cell, and \( l \) indicates the three dimensional lattice vector. In what follows we will also use a single composite index \( a = (a, s, l) \) to indicate Cartesian index, atom index and lattice vector together. Moreover, in general, we will use bold letters to indicate also other quantities in component-free notation.

In order to take into account quantum effects and anharmonicity at nonperturbative level, we use the stochastic self-consistent harmonic approximation (SSCHA)\textsuperscript{8,9,24}. For a given temperature \( T \), the method allows to find an approximate estimation for \( F(R^{a_s}(l)) \), the free energy of the crystal as the function of the average atomic position \( \mathcal{R}^{a_s}(l) \) (the centroids). For a given centroid \( \mathcal{R} \), the SSCHA free energy is obtained through an auxiliary quadratic Hamiltonian, the SSCHA Hamiltonian \( \mathcal{H}_R \). In a displacive second-order phase transition, at high temperature the free energy has minimum in a high symmetry configuration \( \mathcal{R}_n \), but, on lowering temperature, \( \mathcal{R}_{hs} \) becomes a saddle point at the transition temperature \( T_c \). Therefore, the free energy Hessian evaluated in \( \mathcal{R}_{hs} \), \( \partial^2 F/\partial \mathcal{R} \partial \mathcal{R} \bigg|_{\mathcal{R}_{hs}} \), at high temperature is positive definite but it develops one or multiple negative eigendirections at \( T_c \). The SSCHA free energy Hessian in a centroid \( \mathcal{R} \) can be computed by using the analytic formula (in component-free notation)\textsuperscript{24}

\[
\frac{\partial^2 F}{\partial \mathcal{R} \partial \mathcal{R}} = \Phi + (\Phi \Lambda(0) \left[ I - (\Phi \Lambda(0))^2 \right]^{-1} \Phi, \tag{2}
\]

with

\[
\Phi = \left\langle \frac{\partial^2 V}{\partial \mathcal{R} \partial \mathcal{R}} \right\rangle_{\rho H_R}, \tag{3}
\]

\[
\Phi = \left\langle \frac{\partial^4 V}{\partial \mathcal{R} \partial \mathcal{R} \partial \mathcal{R} \partial \mathcal{R}} \right\rangle_{\rho H_R}, \tag{4}
\]

where the averages are with respect to the density matrix of the SSCHA Hamiltonian \( \mathcal{H}_R \), i.e. \( \rho H_R = e^{-\beta H_R}/\text{tr} \left[ e^{-\beta H_R} \right], \) and \( \beta = (k_b T)^{-1} \) where \( k_b \) is the Boltzmann constant. In Eq. (2) the value at \( z = 0 \) of the 4th-order tensor \( \Lambda(z) \) is used. For a generic complex number \( z \) it is defined, in components, by

\[
\Lambda^{abcd}(z) = -\frac{1}{2} \sum_{\mu \nu} F(z, \omega_{\mu}, \omega_{\nu}) \times \sqrt{\frac{\hbar}{2 M_a \omega_{\mu}}} e^a_{\mu} \sqrt{\frac{\hbar}{2 M_b \omega_{\nu}}} e^b_{\nu} \sqrt{\frac{\hbar}{2 M_c \omega_{\mu}}} e^c_{\mu} \sqrt{\frac{\hbar}{2 M_d \omega_{\nu}}} e^d_{\nu}, \tag{4}
\]

with \( M_a \) the mass of the atom \( a, \omega_{\mu}^2 \) and \( e^a_{\mu} \) eigenvalues and corresponding eigenvectors of \( D^{(S)} \) \( = \Phi_{ab}/\sqrt{M_a M_b} \), respectively, and

\[
F(z, \omega_{\nu}, \omega_{\mu}) = \frac{2}{\hbar} \left[ (\omega_{\mu} + \omega_{\nu})(1 + n_b(\omega_{\mu}) + n_b(\omega_{\nu})) + (\omega_{\mu} - \omega_{\nu})^2 - z^2 \right] \left[ n_b(\omega_{\mu}) - n_b(\omega_{\nu}) \right] \left[ (\omega_{\mu} - \omega_{\nu})^2 - z^2 \right] \tag{5}
\]

where \( n_b(\omega) = 1/(e^{\beta \omega} - 1) \) is the bosonic occupation number. Evaluating through Eq. (2) the free energy Hessian in \( \mathcal{R}_{hs} \) and studying its spectrum as a function of temperature, we can predict the occurrence of a displacive second-order phase transition and estimate the relative \( T_c \). In particular, since we are considering a crystal, we take advantage of lattice periodicity and we Fourier transform the free energy Hessian with respect to the lattice indexes. Therefore, since there are 2 atoms in the unit cell of PbTe and SnTe, we actually calculate the eigenvalues \( \lambda_n^2(q) \) of the two dimensional \( 6 \times 6 \) square matrix \( \partial^2 F/\partial \mathcal{R} \partial \mathcal{R}(\mathcal{R} - \mathcal{R}_0)(q) \) in different points \( q \) of
the Brillouin zone. The evaluation of the magnitude of the different term of Eq. (2) is discussed in appendix A.

As showed in Ref. 24, in the context of the SCHA it is possible to formulate an ansatz in order to give an approximate expression to the one-phonon Green function $G(z)$ for the variable $\sqrt{M_0(R^0 - R^0_\infty)}$

$$G^{-1}(z) = z^2 \mathbb{1} - M^{-\frac{1}{2}} \Phi M^{-\frac{1}{2}} - \Sigma(z),$$

(6)

where $G^{-1}(0) = -D^{(F)}$, $D^{(F)} = \frac{1}{\sqrt{M_0 M_b}} \partial^2 \tilde{F}/\partial r_a \partial r_c$, and $\Sigma(z)$ is the SSCHA self-energy, given by

$$\Sigma(z) = M^{-\frac{1}{2}} \mathbb{1} \Phi \Lambda(z) \left[ \mathbb{1} - \Phi(z) \right]^{-1} \Phi(z) M^{-\frac{1}{2}},$$

(7)

where $M_{ab} = \delta_{ab} M_a$ is the mass matrix. As shown in appendix A, for the applications considered in the present paper, the static term $\Phi(z)$ is negligible with respect to the identity matrix (or using the appendix A notation, $\langle D4V \rangle$ is negligible). Extending this approximation to the dynamical case reduces the SSCHA self-energy to the so-called bubble self-energy, namely

$$\Pi(z) \approx \Pi^{(B)}(z) = M^{-\frac{1}{2}} \mathbb{1} \Phi \Lambda(z) \Phi^{(3)} M^{-\frac{1}{2}},$$

(8)

We then neglect the mixing between different phonon modes and assume that $\Pi(z)$ is diagonal in the basis of the eigenvectors $e^{(a)}_\mu(q)$ of $\Phi_{\alpha\beta}(q)/\sqrt{M_a M_b}$ where $\Phi_{\alpha\beta}(q)$ is the Fourier transform of $\Phi_{\alpha\beta}$. We then define

$$\Pi_\mu(q, \omega) = \sum_{\alpha\beta} e^{(a)}_\mu(q) \Pi_{\alpha\beta}(q, \omega + i0^+) e^{(b)}_\beta(q)$$

(9)

and $\omega^2_\mu(q)$ are the eigenvalues of the Fourier transform of $D^{(S)}$. The phonon frequencies squared, $\Omega^2_{\mu\nu}$, corrected by the bubble self-energy are then obtained as

$$\Omega^2_{\mu\nu} = \omega^2_\mu(q) + \text{Re} \Pi_\mu(q, \omega, \mu)(q)$$

(10)

In studying the response of a lattice to neutron scattering we need the one-phonon spectral function. By using Eq. (6) for $G(z)$ we can calculate the cross-section $\sigma(\omega) = -\text{Tr} \text{Im} G(\omega + i0^+)$, whose peaks signal the presence of collective vibrational excitations (phonons) having certain energies, as they can be probed with inelastic scattering experiments (here the chosen normalization factor is such that $\int d\omega \sigma(\omega)$ is equal to the total number of modes). Again, we take advantage of the lattice periodicity and we Fourier transform the interesting quantities with respect to the lattice indices. In particular, we consider the Fourier transform of the SSCHA self-energy, $\Pi_{\alpha\beta}(q, z)$. Neglecting the mixing between different modes, the cross section is then given by

$$\sigma(q, \omega) = \frac{1}{\pi} \sum_\mu \frac{-\omega \text{Im}\Pi_\mu(q, \omega)}{\omega^2 - \omega^2_\mu(q) - \text{Re} \Pi_\mu(q, \omega^2) + (\text{Im}\Pi_\mu(q, \omega))^2}.$$

(11)

III. METHODS

We perform density-functional theory (DFT) calculations using the QUANTUM-ESPRESSO package. For both systems the exchange-correlation interaction is treated with the Perdew-Burke-Ernzerhof (PBE) generalized gradient approximation. To describe the interaction between electrons and ions, we use norm-conserving pseudopotentials for PbTe, and PAW pseudopotentials for SnTe. In both cases the semi-core 4d states in valence are included for Te and Sn, and 5d for Pb. Electronic wave functions are expanded in a plane-wave basis with kinetic energy cutoffs of 65 Ry and 28 Ry for scalar relativistic pseudopotentials, for PbTe and SnTe respectively. Integrations over the Brillouin zone (BZ) are performed using a uniform grid of $8 \times 8 \times 8$ k-points for PbTe, and a denser grid of $12 \times 12 \times 12$ k for SnTe. Particular care must be taken in converging SnTe with respect to k-points as the depth of the potential well as a function of phonon displacements is strongly dependent on the sampling. Previous calculations carried out with smaller samplings are found to be underconverged (see appendix B for convergence tests). Born effective charges calculated via density-functional perturbation theory (DFPT) are included for PbTe calculations only. For both PbTe and SnTe we consider the high temperature rock-salt structure, and the PBE optimized lattice parameters of 6.55 Å and 6.42 Å respectively. The effect of the thermal expansion is discussed in Appendix C.

Harmonic phonon frequencies are calculated within the DFPT as implemented in QUANTUM-ESPRESSO. We investigate the $2 \times 2 \times 2$, and then the $4 \times 4 \times 4$ q-point grids for both systems. Fourier interpolation is used to obtain the phonon dispersion along high symmetry lines.

To calculate the anharmonic renormalized phonons we use the SSCHA. The trial Hamiltonians is minimized in a supercell. This minimization process requires the energies and forces acting on a supercell for a set of random configurations generated by the trial density matrix. Those elements have been calculated on $2 \times 2 \times 2$ and $4 \times 4 \times 4$ supercells using the same parameters for the harmonic DFPT calculations. The number of random configurations we use is of the order of one thousand. The difference between harmonic and anharmonic dynamical matrices is interpolated to a $14 \times 14 \times 14$ supercell for SnTe. For PbTe no interpolation is needed since the $4 \times 4 \times 4$ supercell is converged and adequate to describe the experimental results.

IV. RESULTS & DISCUSSION

A. Harmonic phonon dispersion

In ferroelectrics and thermoelectrics the phonon spectra strongly depends on the volume used in the
FIG. 1. (Color online) Harmonic phonons dispersion of PbTe and SnTe for different lattice constants. For the experimental lattice parameter (red lines) both systems do not present negative frequencies. For the PBE optimized lattice parameter (blue lines), the SnTe phonon spectra has negative frequencies on Γ, indicating a structural instability. The TO modes at Γ exhibit an strong dependence on the volume.

In order to determine if the SSCHA approximation can describe phonon satellites and to investigate the occurrence of the avoided-crossing, the phonon self-energy is calculated, we first investigate the dependence of the harmonic phonon spectra on the lattice parameters. For this reason in Fig. (1) we calculate harmonic phonon dispersions within DFPT for PbTe and SnTe using (i) the zero temperature PBE theoretical lattice parameter and (ii) the experimental\textsuperscript{31,32} ones \(a_{\text{exp}} = 6.46\,\text{Å} \) and \(6.32\,\text{Å}\), respectively. In both case the experimental parameter is smaller then the theoretical one, as if the system were experiencing a finite pressure. As expected, the results are strongly volume dependent. In the case of PbTe, the use of the experimental lattice parameter hardens all the phonons, but the hardening is particularly large for the transverse optical (TO) mode at zone center that is shifted from 3.17 meV to 6.19 meV. However in both cases, the harmonic phonon frequencies are positive and no structural instability is detected in PbTe, in agreement with experiments. PbTe is usually referred as an incipient ferroelectric because of the softness of the TO phonon mode. It is important to underline that in the case of PbTe the experiments\textsuperscript{14,15} find a clear LO/TO splitting. We thus included this effect in our harmonic calculation.

At ambient pressure SnTe undergoes a phase transition in the 30 — 100 K temperature range. At low temperature the crystal symmetry changes from cubic (Fm\(3m\)) to rhombohedral (R\(3m\)). The distortion is a displaceive phase transition involving a small dimerization in the unit cell\textsuperscript{33}. The distortion is compatible with a phonon instability at zone center. Real samples of SnTe are non stoichiometric and the ferroelectric transition temperature strongly depends on the number of holes present in the system. It is approximately 100 K for hole concentrations of the order of \(1 \times 10^{20} \,\text{cm}^{-3}\) and decreases to approximately 30 K for ten times larger hole concentrations. At these large doping no LO/TO splitting is expected, so we neglect it in the simulation.

The dependence of the harmonic calculation on volume is stronger in the case of SnTe. The ferroelectric transition (imaginary TO phonon at zone center) is present when using the more expanded theoretical PBE volume while it disappears if the experimental volume is used. This again underlines the critical role of the volume used in the calculation of phonon spectra in ferroelectrics and thermoelectrics\textsuperscript{34}. Finally, we also investigated the role of hole doping by using the virtual crystal approximation in appendix D. In the rest of the paper we consider the PBE optimized lattice parameter at \(T = 0\,\text{K}\) in all calculations. For SnTe, as we are interested in the temperatures below 100 K, we neglect the effects of the thermal expansion.

B. Anharmonic phonons

1. Lead telluride (PbTe)

In Fig.(2) we compare the PbTe phonon dispersion curves \(\Omega_{q\nu}\) at 300 K obtained by the SSCHA using Eq. 10 to INS experimental data obtained by Cochran et al.\textsuperscript{14}. Our calculated curves are in good agreement with experimental results. We obtain a higher value than the experiments for the TO modes at zone center is consistent to newer observations of a double peak in this region\textsuperscript{15}. Previous calculations\textsuperscript{22} found a good agreement for the lower energy TO mode at Γ, however not obtaining a good description of the high energy phonon branches. More recent INS measurements\textsuperscript{15} have suggested the presence of a strong temperature dependent phonon satellite close to Γ originated from the TO mode. Furthermore, as the temperature increases, an avoided crossing between LA and TO phonon bands along the ΓX direction is reported at \(T > 300\,\text{K}\).

In order to determine if the SSCHA approximation can describe phonon satellites and to investigate the occurrence of the avoided-crossing, the phonon self-energy is
calculated performing Fourier interpolation over a denser $40 \times 40 \times 40$ phonon momentum grid. Fig.(3) shows our calculated anharmonic phonon dispersion versus the neutron cross section $\sigma(\omega)$ computed using Eq. 11 for PbTe.

We also show, with pink dots, the Energy of the TO phonon and of its satellite as measured in INS experiments detailed in Ref.15. Both the satellite and the crossing of LA and TO bands at $300 \text{K}$ are well described by our methodology. Moreover, the energies of the TO peaks at $\Gamma$ obtained by the SSCHA are compatible with the observed values. The presence of these features were also investigated in the literature using different methods. In particular, non-perturbative methods such as the temperature-dependent effective potential technique (TDEP)$^{6,19}$ were able to obtain similar results.

2. Tin telluride (SnTe)

Calculations for SnTe are reported in Fig.(4) where the comparison between the phonon spectra obtained via SSCHA at $T = 100 \text{K}$ and recent IXS experiments$^{17}$ at $T = 75 \text{K}$ is shown. Even, if this system was studied theoretically before, using methods such as the TDEP and self-consistent ab-initio lattice dynamics (SCAILD)$^{7,17}$, the calculations focused on higher temperatures. Our calculated anharmonic dispersion curves present the main features of the experimental data for all investigated high symmetry directions along the BZ. Overall, we find a good agreement with experimental data. Furthermore we also investigate the neutron cross section $\sigma(\omega)$ throughout the Brillouin zone but we did not detect any phonon-phonon satellite.

In order to study the second order structural phase transition in SnTe we evaluate the energy squared of the TO modes at $\Gamma$ as a function of temperature $T$. Our data with the inclusion of anharmonicity are consistent with a ferroelectric transition at $\approx 23 \text{K}$. However, this value should be taken with care as the theoretical calculations are limited by the error in the knowledge of the exchange correlation functional that leads to a big variation in the equilibrium volume. On the other hand, experimentally, the transition temperature of SnTe is strongly dependent on sample doping, varying from 0 $\text{K}$ to around 120 $\text{K}$ for different carriers concentrations. Fig.(5) compares our results for the energy squared of the TO mode with recent IXS$^{17}$ experiments.

V. CONCLUSIONS

We applied a novel technique$^{24}$ based on the stochastic self consistent harmonic approximation capable of investigating phase transitions via the calculation of the
FIG. 4. (Color online) Harmonic (dashed lines) and anharmonic (solid lines) phonon dispersion relations of SnTe at 100 K (red lines) compared with IXS experiments at 75 K (black dots). The anharmonic phonon dispersion ($\Omega_{qu}$) is obtained from Eq. (10) and includes the contribution from the bubble self-energy.

Hessian of the free energy. We have studied the temperature dependent anharmonic phonon spectra of PbTe and SnTe. We found a strong dependence of vibrational properties on the exchange-correlation functional used in the calculation and on the corresponding equilibrium volume. By using the PBE functional with the theoretical equilibrium volume, we find very good agreement with experimental INS spectra. The SSCHA is not only capable of describing single particle spectra, but also manybody features like phonon satellites are correctly explained. Finally, we describe the occurrence ferroelectric transition in SnTe from the high-T Fm$3\overline{m}$ structure to the low-T R$3m$ one. The value of the ferroelectric critical temperature is found to be strongly dependent in the volume used in the calculations, consequently, on the exchange-correlation functional.

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Appendix A: Magnitude of $\langle D3V \rangle$ and $\langle D4V \rangle$ terms

In order to have a better visualization of the different terms of Eq.(2) one can rewrite it as:

$$\frac{\partial^2 F}{\partial R \partial R} = \Phi + \Phi \Lambda(0) \Phi + \Phi \Lambda(0) \Theta \Lambda(0) \Phi,$$

where

$$\Theta = \left[ I - \Phi \Lambda(0) \right]^{-1} \Phi.$$ 

For simplicity, we define

$$\langle D3V \rangle = \Phi \Lambda(0) \Phi$$

$$\langle D4V \rangle = \Phi \Lambda(0) \Theta \Lambda(0) \Phi$$

To investigate the different terms in Eq.(A1), and in particular the mutual role of $\langle D3V \rangle$ and $\langle D4V \rangle$ we perform SSCHA runs, on a $2 \times 2 \times 2$ supercell, for different temperatures; 300 and 600 K for PbTe, and 50 and 100 K for SnTe.
FIG. 6. (Color online) Anharmonic phonon dispersion curves for PbTe at 300 K and 600 K. Orange lines denote calculations neglecting the bubble and superior order terms, \( \omega_\mu(q) \); black dashed lines represent the calculations including \( \langle D^3 V \rangle \), while red lines includes the full expression, \( \langle D^3 V \rangle + \langle D^4 V \rangle \), \( \omega_\mu(q) \). The phonon frequencies are obtained using Eq.(10) in the static limit, namely by using \( \Pi_\mu(q, 0) \).

Fig. 7 (Color online) SSCHA runs for SnTe at 50 K and 100 K. Orange lines denote calculations neglecting the bubble and superior order terms, \( \omega_\mu(q) \); black dashed lines represent the calculations including \( \langle D^3 V \rangle \), while red lines includes the full expression, \( \langle D^3 V \rangle + \langle D^4 V \rangle \), \( \omega_\mu(q) \). The phonon frequencies are obtained using Eq.(10) in the static limit, namely by using \( \Pi_\mu(q, 0) \).

Appendix B: k-points sampling in SnTe

The convergence of DFT calculations for SnTe is quite tricky. Even if the difference on the total energy of the Fm\(\bar{3}\)m structure between the \(8 \times 8 \times 8\) and \(20 \times 20 \times 20\) k-point grids is of the order of 0.8 meV/atom, upon distortion towards the R\(\bar{3}\)m structure, the depth of the potential well differs of 1.55 meV/cell. Thus, using a smaller k-point grid, as done previous calculations\(^2\), substantially overestimates the ferroelectric instability, as shown in Fig. 8. For this reason we used a converged \(12 \times 12 \times 12\) k-point grid in our SnTe calculations.

Appendix C: Evaluation of PbTe’s thermal expansion at 600K

To evaluate the effects of thermal expansion in our calculations for PbTe at 600 K, we perform several SSCHA runs for different volumes using a \(2 \times 2 \times 2\) supercell. We
add the vibrational free energy to the BO total energy to construct an energy vs lattice parameter curve. By finding the minimum of this curve we obtain a lattice parameter of 6.642 Å for PbTe at 600 K using the PBE functional in our calculations. With this new lattice constant, we then compute the anharmonic phonon dispersion as before. Fig.(9) shows the phonon spectra for the two lattice parameters (PBE $T = 0 \text{ K}$ and $T = 600 \text{ K}$) considering just $\langle D3V \rangle$ and the case $\langle D3V \rangle + \langle D4V \rangle$. The phonon frequencies shift towards smaller values in relation to the PBE at 0 K as illustrated in Fig.(7). The shift is not significant for the $\langle D3V \rangle + \langle D4V \rangle$ case, whereas for the other, therm beyond the bubble gain more importance.

Appendix D: Born effective charges and doping

The fact that both undoped PbTe and SnTe do not exist stoichiometric raises questions on how those systems should be simulated. For PbTe, since the typical doping in quite small and the LO-TO splitting is quite pronounced, the inclusion of the Born effective charges is a natural choice. The effective charges for PbTe are calculated via DFPT and a posteriori added to the SS-CHA dynamical matrices. On the other hand, on SnTe the effects of doping on the phonon dispersion are way more significant. Undoped SnTe should be ferroelectric, however its nature depends on the carriers concentration. Typically SnTe is heavily hole doped. As hole concentration increases, the transition temperature decreases down to a point in which the system remains cubic, even at low temperatures, hence losing the ferroelectric phase. In order to tackle this problem, we compared the harmonic phonon dispersion for doped and undoped SnTe on a $4 \times 4 \times 4$ supercell. We have not included the Born effective charges as they would be screened by doping, and we used $n_h = 3.23 \times 10^{20} \text{ cm}^{-3}$ from ref. as the carrier concentration. Fig.(10) shows that the instability remains at this doping level, and the phonon dispersion along high symmetry directions are almost unaffected within the BZ.

Appendix E: Empirical potential calculations

In this work, in order to investigate supercell size effects on the phonons modes, specially at the zone center,
in addition to the full ab-initio calculations presented in the main text, we made use of a model potential based on the formulation developed by Marianetti et al.\textsuperscript{35,36} In the main text, we made use of a model potential based on the main text, we made use of a model potential based on the formulation developed by Marianetti et al.\textsuperscript{35,36}

The potential has the form:

$$V(R) = \frac{1}{2} \sum_{s \alpha} \phi_{ab} u^a_{s \alpha} u^b + V^{(3)}_A(u) + V^{(4)}_A(u),$$  \hspace{1cm} (E1)$$

where $u = R - R_{(0)}$, $R_{(0)}$ corresponding to the equilibrium configuration on the rock-salt structure. The harmonic matrices $\phi_{ab}$ were calculated using the same parameters as for the DFPT calculations described in the main text. Anharmonic terms $V^{(3)}_A$ and $V^{(4)}_A$ are defined as:

$$V^{(3)}_A(u) = p_3 \sum_{s = 1}^{N_s} \sum_{\alpha = x,y,z} [A^{x}_{s,\alpha+} + A^{x}_{s,\alpha-}]$$  \hspace{1cm} (E2)$$

and

$$V^{(4)}_A(u) = p_4 \sum_{s = 1}^{N_s} \sum_{\alpha = x,y,z} \left[ A^{x}_{s,\alpha+} \left( (E^{(1)}_{s,\alpha+})^2 + (E^{(2)}_{s,\alpha+})^2 \right) 
+ A^{x}_{s,\alpha-} \left( (E^{(1)}_{s,\alpha-})^2 + (E^{(2)}_{s,\alpha-})^2 \right) \right]$$  \hspace{1cm} (E3)$$

where, for example

$$A_{s,x} = \frac{1}{\sqrt{2}} \left( u^s_{x}(s),x - u^s_{x}(s) \right)$$

$$E^{(1)}_{s,x} = \frac{1}{\sqrt{2}} \left( u^s_{x}(s),y - u^s_{x}(s) \right)$$

$$E^{(2)}_{s,x} = \frac{1}{\sqrt{2}} \left( u^s_{x}(s),z - u^s_{x}(s) \right).$$  \hspace{1cm} (E4)$$

The variables $x_{s}(s)$ and $x_{s}(s)$ represent the nearest-neighbour of atom $s$, along the cartesian direction $+x$ and $-x$, respectively. For the other cartesian directions, $+y$ and $+z$, we generalized this notation. The quantity $u$ is the displacement from the equilibrium position.

The potentials for both systems were defined by fitting the parameters $p_3$, $p_4$, and $p_{4x}$ to ab-initio forces calculated for one thousand random atomic configurations generated in the first principles SSCHA calculation. For PbTe we used a combination of configurations generated at $300\,K$ and $600\,K$, resulting on the coefficients $p_3 = 2.99\,eV/(A)^3$, $p_4 = 4.17\,eV/(A)^4$, and $p_{4x} = -1.32\,eV/(A)^4$. For SnTe we used configurations generated at $100\,K$ obtaining $p_3 = 2.51\,eV/(A)^3$ and $p_4 = 6.18\,eV/(A)^4$, in this case $p_{4x}$ was neglected since its contribution was not relevant.

Fig.(11) and Fig.(12) compare the phonon dispersion on a $4 \times 4 \times 4$ supercell calculated ab-initio and using the empirical potential for PbTe ($300\,K$) and SnTe ($100\,K$), respectively. For PbTe the larger difference is at zone center, this may be due to the fact that we used random configurations generated on a broader range of temperatures. However, this is not a problem in order to study the convergence of the TO modes using the empirical potential for different supercell sizes. For SnTe, since we generated our random configurations at $100\,K$, one may expect a better agreement.

Fig.(15) presents the convergence tests regarding the TO modes of PbTe and SnTe. For the first compound, the difference between the $2 \times 2 \times 2$ and $4 \times 4 \times 4$ is small for the points which are included exactly by using the $2 \times 2 \times 2$ supercell (X,Γ,L), as shown in Fig.(13). However, as stated before, we used the $4 \times 4 \times 4$ supercell in order to include more points along on the BZ and, as a consequence, to describe more accurately the phonon dispersion of PbTe. As mentioned on the main text, for SnTe we considered at least a $4 \times 4 \times 4$ supercell in our ab-initio calculations, since the $2 \times 2 \times 2$ does not seem
to be sufficient. In order to test the convergence we explored the model potential on the $4 \times 4 \times 4$ and also on

the $5 \times 5 \times 5$ supercells. In this case, we compared just the SSCHA runs without including extra terms. Fig.(14) shows the phonon dispersion for the $4 \times 4 \times 4$ and $5 \times 5 \times 5$ supercell. It is necessary to emphasize that the latter presents some wiggles due to the Fourier transform, so the negative energies are not physical, just an interpolation artifact.

As can be viewed in Fig.(15), the difference between the TO modes using different supercells is not significant for our purposes.
FIG. 14. (Color online) Comparison between SnTe anharmonic phonon spectra using a $4 \times 4 \times 4$ (red) and $5 \times 5 \times 5$ (purple) supercell. In this case the bubble and higher order terms are neglected on our calculations. The phonon frequencies ($\omega_{\mu}(q)$) are obtained using Eq.(10).

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FIG. 15. (Color online) On the TO modes in function of $T$ for PbTe (top) and SnTe (bottom) for different supercell sizes. The bubble contribution was taken on for supercells up to the $4 \times 4 \times 4$. The phonon frequencies ($\omega_{\mu}(q)$) are obtained using Eq. (10) in the static limit, namely by using $\Pi_{\mu}(q, 0)$.