Current rectifying and resistive switching in high density BiFeO$_3$ nanocapacitor arrays on Nb-SrTiO$_3$ substrates

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Ultrahigh density well-registered oxide nanocapacitors are very essential for large scale integrated microelectronic devices. We report the fabrication of well-ordered multiferroic BiFeO$_3$ nanocapacitor arrays by a combination of pulsed laser deposition (PLD) method and anodic aluminum oxide (AAO) template method. The capacitor cells consist of BiFeO$_3$/SrRuO$_3$ (BFO/SRO) heterostructural nanodots on conductive Nb-doped SrTiO$_3$ (Nb-STO) substrates with a lateral size of ~60 nm. These capacitors also show reversible polarization domain structures, and well-established piezoresponse hysteresis loops. Moreover, apparent current-rectification and resistive switching behaviors were identified in these nanocapacitor cells using conductive-AFM technique, which are attributed to the polarization modulated $p$-$n$ junctions. These make it possible to utilize these nanocapacitors in high-density (>100 Gbit/inch$^2$) nonvolatile memories and other oxide nanoelectronic devices.

BiFeO$_3$ (BFO) has attracted intensive attention in the past decades due to its well-known room temperature multiferroicity, in addition to excellent ferroelectric, magnetoelectric, photovoltaic, and electromechanical properties, offering a series of promising applications in high density memory, photovoltaic, electromechanical, and spintronic devices$^{1-3}$. With the current trends of high integration and miniaturization in semiconductor industry, nanoscale multiferroic/magnetoelectric materials (e.g. BFO) as cutting-edge nano-electronic research field have been receiving more and more attention.

It was reported that BFO nanodots exhibit various unique properties inaccessible otherwise$^{6-10}$. For instance, BFO nanoparticles show strong size-dependent magnetic and photocatalytic properties due to the size confinement$^{7-8}$. In free-standing tetragonal-like BFO nano-islands, an unexpected shape memory behavior in association with a martensitic-like phase transformation was observed, owing to the release of substrate clamping$^{11}$. Isolated nanocapacitor structures can also greatly enhance the stability of the ferroelastic domain switching in BFO, enabling the electric control of antiferromagnetism$^{12}$. However, for large scale integration devices, high density capacitor array is very essential, nevertheless up to now there have been yet very few reports addressing both their fabrication process and nanoelectronic properties$^{13}$. On the other hand, the electrical properties of nanocapacitor also critically depend on their electrodes, which not only affect the uniformity of electric field distribution inside the capacitors, but also change their piezoelectric and transportation behaviors via interfacial barrier modulation$^{14,15}$. For instance, we have demonstrated in a previous work that a unique complex domain structure along with an enhanced resistive switching behavior, can be introduced by self-assembled Bi$_2$O$_3$ nanoisland top electrodes in ultrathin BFO film-based nanocapacitors$^{15}$.

In this work, we have developed well-ordered BFO nanocapacitor arrays with SrRuO$_3$ (SRO) top electrodes on conductive Nb-SrTiO$_3$ (Nb-STO) substrates by the anodic alumina (AAO) template assisted method. We have selected Nb-STO as bottom electrode as it has been reported to significantly improve the resistive switching behavior of ultrathin ferroelectric BaTiO$_3$ film$^{16}$. The nanocapacitor arrays have well-epitaxial structure, showing apparent ferroelectric polarization and interesting current rectifying resistive switching characteristics. They are promising for applications in ultrahigh density recording devices.
Results and discussion

To construct the nanocapacitor array, we have deposited BFO/SRO heterostructured nanodots on conductive Nb-STO substrates which serve as bottom electrodes. In brief, the BFO/SRO heterostructured nanodots were sequentially grown on a conductive Nb-doped (100) SrTiO₃ (Nb-STO) substrate through an AAO template by pulsed laser deposition (PLD). The fabrication process is illustrated in Fig. 1(a), the details of which will be presented in the method section, and further information can also be found in our previous reports. As shown in Fig. 1(c), the BFO nanodots exhibit an average lateral size of 60 nm, and a dot-dot distance of 120 nm. The XRD spectrum shows a (001)-orientated BFO/SRO heterostructure on Nb-STO, as reflected by the (00$l$) diffraction peaks shown in Fig. 1(d).

The Nb-STO/BFO/SRO nanoscale features were also examined by the cross-section TEM observations shown in Fig. 2. The cross-section image demonstrates that each cell is composed of well-epitaxial BFO thin layer of 10 nm in thickness, covered with a SRO capping layer as top electrode. It is also worthy of mention that the interface between BFO and SRO nanodots is not very flat, which is limited by our fabrication method. The individual layers were also checked by Energy Dispersive X-Ray Spectroscopic Analysis (EDX). From the transmission electron microscopy (TEM) cross-section images, it is seen that both BFO and SRO show well-established single crystalline with the same orientation as the substrate. The epitaxial quality was further examined by selected area electron diffraction (SAED) along the <010> direction (Fig. 2(c)), where apparent diffraction spots of STO/SRO and BFO, along with minor amount of impurity phase, can be identified. It should be mentioned here that, the diffraction spots of SRO and STO are well overlapped, as their lattice parameters are very close. By carefully examining the reciprocal lattices from the STO and BFO, we can derive a large BFO c/a ratio of 1.14, corresponding to an out-of-plane lattice parameter $c \sim 4.45$ Å, close to that of tetragonal BFO nanostructure on LaAlO₃ substrate reported by Zhang et al ($c = 4.65$ Å). We have checked carefully the lattice space and have found that the c-parameter is not very uniform spatially. Most areas show a small c-lattice space $c \sim 4.5$ Å, while there are some locations exhibiting smaller $c \sim 4.0$ to 4.2 Å. The diffraction spots can well reflect the average of the lattice spaces, consistent with our HRTEM image analysis. We also notice that the angle between the $a$ and $c$ is 89°–90°, slightly deviated from the rectangular angle. Therefore, it is safe to conclude that the structure is monoclinic or most likely pseudotetragonal, similar to reported results for BFO/SRO/Nb-STO by Chu et al.

Figure 1 | Fabrication details for the nanocapacitor arrays. (a) Schematic flow chart illustrating the fabrication procedure for the BFO-SRO nanocapacitor arrays on Nb-STO substrate; (b, c) three-dimensional topographic image for the AAO mask (b) and BFO/SRO nanodots (c); (d) XRD diffraction pattern for the as-deposited SRO/BFO/Nb-STO nanodot heterostructures.

Figure 2 | Cross-section TEM images for the SRO/BFO/Nb-STO nanodot heterostructure. (a) Relative smaller magnification image as a overview for a nanocapacitor structure, and (b) larger magnification image; (c) selected area electron diffraction (SAED) along the <010> direction, showing diffraction spots of STO, SRO and BFO, along with minor amount of impurity phase.
While the nonuniformity in lattice space may be related to the defects such as dislocations, which can partially relax the local strains, further study is still needed to throw light on the reason for the large lattice distortion in our BFO nanostructures.

To characterize the ferroelectric properties of the nanodots, vertical piezoresponse force microscopy (VPFM) measurements were performed and the results are highlighted in Fig. 3. Fig. 3(a) shows the AFM topography, piezoresponse amplitude- and phase-contrast micrographs for the nanodot arrays. The bright- and dark-contrasts in the phase micrographs correspond to the down-polarization ($P_{\text{down}}$) and up-polarization ($P_{\text{up}}$) states, respectively, while the contrast in amplitude piezoresponse is related to the magnitude of the piezoelectric signal. To show the polarization reversal status, the BFO-SRO nanodot array was first electrically poled by applying an external scanning bias at a pre-designed area during the scan, in which the middle area was poled downwards by a reverse voltage of $-6\, \text{V}$, and the rest area upwards with a positive voltage of $+6\, \text{V}$. From the phase-contrast micrograph, we can observe completely different dark- and bright-contrast area for the dots of different polarization orientations, indicating that the polarizations of the nanodots are reversible under applied electric voltages. From the piezoresponse amplitude-contrast image, it was found that the amplitude for the downward polarization is slightly smaller than that of upward, exhibiting some extent of preferred polarization orientation. In between the two different polarization regions, there are some dots exhibit low piezoelectric amplitude, which may be correspondent to those in domain border region. It is known that for most reports, the domain wall width in BFO thin films is a few nanometers\textsuperscript{15,19}. However, for our BFO nanodots, the domain structure and configuration can be quite different. The free boundary of nanodots imposes additional mechanical and electric boundary conditions which make the domain structure of nanodots much more complicated than that for thin films. For example, one can observe upward, downward, bubble-like, vortex-like, and stripe-like domain patterns in nanodots. In these cases, the domain wall width, if definable, may be much wider than those in thin films. Therefore, we can only see a border region in our nanodot array, instead of a sharp boundary. To demonstrate the reversibility of an individual dot, we applied a pulsed voltage of $\pm 6\, \text{V}$ on a selected dot, which produces apparent different contrasts for the two different pulsed voltages, as shown in Fig. 3(b). This confirms that the isolated dot is switchable in polarization by applying a pulsed voltage.

To examine the local electric properties of the nanodots, we measured the piezoelectric hysteresis loops on a single nanocapacitor. The piezoresponse phase-voltage hysteresis and the butterfly-like amplitude-voltage loops are displayed in Fig. 4(a) and (b). At a low bias of $\sim 1\, \text{V}$ (not shown here), both the amplitude and phase remain rather stable, indicating no reversal process. Once the bias voltage increases beyond $3\, \text{V}$, the switching becomes apparent, producing a well-developed butterfly amplitude loop and a square phase hysteresis loops at a bias of $4\, \text{V}$. The two asymmetric coercive fields $V_{+}= 1.32\, \text{V}$ and $V_{-}= -2.16\, \text{V}$ can be identified, indicating that the polarization reverse is non-symmetric, as also confirmed by the as-grown states. This may be due to the built-in fields from the work-function difference between the top/bottom electrodes and BFO. From the band structure analysis, we have found that the SRO and NSTO have their work-functions of around $5.2\, \text{eV}$ and $4.08\, \text{eV}$, respectively\textsuperscript{21,22}, which produces an overall theoretical built-in voltage of $1.12\, \text{eV}$. This breaks the equivalence of two polarization states, and provides a strong tendency to align the domains to a preferred orientation. In additional, oxygen vacancies adjacent the

![Figure 3](https://www.nature.com/scientificreports)

**Figure 3** | Piezoresponse images for the polarization reversal process in the nanocapacitor arrays. (a) Topological, and piezoresponse amplitude and phase images for the nanocapacitor array, in which in the middle square area was poled downwards (with a bias voltage of $-6\, \text{V}$) while the rest part was upwards with a voltage of $+6\, \text{V}$; (b) the piezoresponse phase images illustrating the polarization reversal for a selected nanocapacitor dot, which was poled upwards and then downwards using bias of $\pm 6\, \text{V}$, respectively.
top electrode introduced during the deposition process may contribute to the observed asymmetric polarization states.

To further examine the resistive property, we look into the local current-voltage ($I-V$) characteristics by C-AFM on a single BFO nanodot. The schematic structure of the device is depicted in Fig. 5(a). One observed a non-degenerated semiconductor, the Fermi level is at least 3 eV below the conduction band diagram with Nb-STO, while Nb-STO has high conductivity and an electron affinity of 3.3 eV, indicating the coexistence of semiconductor diode relation below the middle of the energy band gap, so the work-function of BFO is deduced to be (3.3 + 1.4 + y) eV, where $y$ is another small value. The work-function of SRO is 4.7 eV, and $V_{bi}(SRO/BFO) = 5.2 - (4.7 + y) - 0.5$ V, and $V_{bi}(BFO/Nb-STO) = (4.7 + y) - (4.08 + y) - 0.62$ V. The two built-in voltages are aligned along the same direction, leading to a big total built-in voltage of ~1.1 V, which can account for the apparent asymmetry and large imprint field of 0.84 V in piezoresponse loops shown in Fig. 4.

From the band structure in Fig. 6(a), the calculated barrier height for electrons and holes are 1.32 eV and 1.72 eV, respectively. Typically, if the barrier height for holes is 0.2 eV higher than that for electrons, the hole current will be approximately a factor of 100 smaller than the electron current. Therefore, the conductive behaviors are mainly dominated by the major charge carrier (electrons). The barrier height for the electrons is 1.32 eV, corresponding to a turn-on voltage of 1.32 V for an ideal $p-n$ junction. From the $I-V$ curve in Fig. 5(b), we can evaluate the two different turn-on voltages of 1.4 V for the HRS and 0.7 V for the low LRS, respectively. This is more or less deviated from the ideal turn-on voltage of 1.32 V, likely due to the band modulation by ferroelectric polarizations.

The observed resistive switching behavior in the SRO/BFO/Nb-STO heterostructures can be accounted for by the ferroelectric polarization modulation on both the width of depletion region and the height of potential barrier at the BFO/Nb-STO interface, as illustrated by our schematic $p-n$ junction model. If no external polarization influences, the depletion width across the BFO/Nb-STO may be considered as a non-degenerated semiconductor, the Fermi level is at least 3 eV below the conduction band diagram. Nb-STO has an energy band gap of 3.2 eV and an electron affinity of 4.08 eV, while BFO has energy band gap of 2.8 eV and an electron affinity of 3.3 eV. BFO forms a staggered energy band diagram with Nb-STO, while Nb-STO has high conductivity and its Fermi level is close to the bottom of the conduction band. For a non-degenerated semiconductor, the Fermi level is at least 3 $kT$ above the energy level of valance band of (E_v) or 3 $kT$ below the energy level of conductive band (E_c). Therefore the work-function of Nb-STO is deduced to be (4 + 3 $kT$ + x) = (4.08 + x) eV, where x is a small value. If no external polarization influences, the depletion width across the BFO/Nb-STO may be considered as a non-degenerated semiconductor. Thus, a $p-n$ junction can be formed at the BFO/Nb-STO interface, which is most probably the reason for the large current rectification behavior.
stay at a certain degree after reaching the dynamic equilibrium state. At downward polarization, the negative majority electron carriers in the n-type Nb-STO are attracted by the positive bound charges and migrate away from the interface, resulting in a decrease in the depletion width, as schematically shown in Fig. 6(b). In contrast, when the polarization is aligned upwards, the negative majority electron carriers in the n-type Nb-STO are repelled by the negative bound charges at the BFO/Nb-STO interface, which increases the depletion width, as shown in Fig. 6(c). The depletion region also induces an energy band bending leading to the change of potential barrier height at the interface, resulting in the variation in turn-on voltages at the two different polarization orientations as exhibited in Fig. 6(b). This agrees well with the experimental values of the turn-on voltage of 0.7 V for LRS and 1.4 V for HRS, respectively (shown Fig. 5(b)). Therefore, the resistive switching behavior in SRO/BFO/Nb-STO heterostructure could be attributed to the modulation of both the depletion width and the potential barrier height by polarization reverse at the BFO/SRO interface. It is also worthy of mention that the SRO/BFO interface can play an important role in the current rectifying behavior. However, for our SRO/BFO nanostructures and heterostructures deposited using the similar parameters, we are not able to obtain such big current rectification ratio (~1000) while it is rather commonly observed for BFO/Nb-STO heterostructures. 21,24 Furthermore, if the p-n junction behavior is from the BFO/SRO interface, it would produce a backward p-n junction I–V behavior instead of the observed forward p-n junction behavior. Therefore, the observed large current rectification is more likely dominated by the BFO/Nb-STO interface.

In summary, well-ordered SRO/BFO/Nb-STO nanocapacitor arrays have been successfully fabricated by PLD in combination with the ultrathin AAO stencil masking method. The BFO nanodots show well-epitaxial tetragonal-like structure with large c/a ratio due to the substrate induced compressive strain. The nanocapacitor arrays show well reversible ferroelectric polarization. Moreover, these BFO nanodots present an apparent resistive switching behavior along with a diode-like rectifying current-voltage characteristic, which are accounted for by a polarization modulated p-n junction model. These results indicate that the nanocapacitor arrays have potential for nano-device applications.

Methods

Fabrication of nanocapctor array. The fabrication procedure of BFO/SRO nanostructures is illustrated in Fig. 1(a). First, the BFO nanodots were epitaxially grown on a conductive 0.7 wt% Nb-doped (100) SrTiO3 (Nb-STO) substrate through a two-step anodization of an AAO template (Fig. 1(b)) by PLD using a KrF excimer laser (λ ~ 248 nm) and an ambient temperature of 500 °C and a low oxygen pressure of 2 Pa. Subsequently, the SRO top electrodes were deposited through the AAO template on top of BFO at the same temperature. Finally, the AAO mask was lifted-off by mechanical method, leaving the well-ordered nanocapacitor arrays as shown in Fig. 1(c). Here, the AAO templates with ~60 nm pore size were fabricated by a two-step anodization of electropolished Al sheets. During the procedure, the first anodization of Al sheet in 0.3 M H2C2O4 solution was conducted for 24 h at 5 °C, then the anodized Al sheet was completely removed in a concentrated acid mixture of H3PO4 and CrO3 (6.0 wt% and 1.8 wt%) at 45 °C for 12 hrs. The second anodization was carried out for 5 mins at 5 °C to get well ordered pores, followed by an etching process in CuCl2 at 10 °C which detached the alumina layer from the Al sheet. In sequence, the barrier layer was removed during the pore widening process with 5 wt% H3PO4 at 35 °C for 30 min. Finally, we obtained ~300 nm-thick AAO membranes that can be transferred to various substrates (e.g. Nb-STO).

Structure and nanoscale electric characterizations. The crystallinities of Nb-STO/BFO/SRO nanocapacitors were characterized by X-ray diffraction (PANalytical X’Pert PRO). The cross-section images were illustrated by high resolution transmission electron microscopy (HRTEM, JOEL-2001). The topology was examined by atomic force microscopy (AFM). The ferroelectric domain structures were probed by piezoresponsce force microscopy (PFM) (Cypher, Asylum Research) using a dual-frequency resonant-tracking technique (DART), and the local I-V curve and resistive switching loops were obtained by fixing conductive atomic microscopy (CAFM) with Pt/Ti coated conductive AFM probes (Nanosensor), where the AFM probe were fixed at a certain point, and then sweeping the dc bias upwards and downwards between bias voltage of ±6 V for certain cycles.

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**Acknowledgments**

The authors would like to thank the Natural Science Foundation of China (Grant Nos. 51031004, 51272078, 51332007, 51431006), the State Key Program for Basic Researches of China (Grant No 2015CB921202), the Project for Guangdong Province Universities and Colleges Pearl River Scholar Funded Scheme (2014), the International Science & Technology Cooperation Platform Program of Guangzhou (No. 20144508016), and the Program for Changjiang Scholars and Innovative Research Team in University of China (Grant No. IRT1243) for financial assistance.

**Author contributions**

L.N.Z. conducted the data acquisition and helped draft the manuscript, Z.X.L. and G.T. participated in the sample fabrication and X.R.D. measurement. F.Y.Z. and W.Z.L. carried out the PFM and CAFM measurement. K.R.H. and Z.Z. contributed to the AAO preparation. S.J.W., X.B.L., M.H.Q. and M.Z. contributed to the data interpretation. J.Y.D. contributed to the TEM observation. X.S.G. & J.M.L. contributed to the data interpretation and manuscript writing. X.S.G. supervised the research.

**Additional information**

Competing financial interests: The authors declare no competing financial interests.

**How to cite this article:** Zhao, L. et al. Current rectifying and resistive switching in high density BiFeO₃ nanocapacitor arrays on Nb-SrTiO₃ substrates. *Sci. Rep.* **5**, 9680; DOI:10.1038/srep09680 (2015).