Advances in the Structural Design of Polyelectrolyte Complex Micelles

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ABSTRACT: Polyelectrolyte complex micelles (PCMs) are a unique class of self-assembled nanoparticles that form with a core of associated polycations and polyanions, microphase-separated from neutral, hydrophilic coronas in aqueous solution. The hydrated nature and structural and chemical versatility make PCMs an attractive system for delivery and for fundamental polymer physics research. By leveraging block copolymer design with controlled self-assembly, fundamental structure–property relationships can be established to tune the size, morphology, and stability of PCMs precisely in pursuit of tailored nanocarriers, ultimately offering storage, protection, transport, and delivery of active ingredients. This perspective highlights recent advances in predictive PCM design, focusing on (i) structure–property relationships to target specific nanoscale dimensions and shapes and (ii) characterization of PCM dynamics primarily using time-resolved scattering techniques. We present several vignettes from these two emerging areas of PCM research and discuss key opportunities for PCM design to advance precision medicine.

1. INTRODUCTION

Controlled self-assembly and compartmentalization on the 1–1000 nm length scale in solution have been longstanding goals in nanotechnology, a field that is beginning to address emerging challenges in energy management,3 green catalysis,1 surfactant compatibilizers,4 and human health.5 Polymeric micelles, which undergo microphase separation, have provided a rich array of hierarchical nanoaggregates that have been widely recognized as leading candidates to address these issues. These nanoparticles allow cargo to be packaged into discrete domains that can withstand inhospitable environments and transport molecules across otherwise impermeable barriers. Micelle assembly is commonly driven either by amphiphilic polymer association in selective solvents or by charged polymer interaction in aqueous solution. Significan...
array of fundamentally unique static and dynamic properties. Polyelectrolyte complex materials can be engineered into polyelectrolyte complex micelles (PCMs) or nanoparticles with a complex core interior and a hydrophilic corona exterior. As shown in Figure 1, PCMs typically employ the coassembly of oppositely charged polymers where at least one polymer in the system has block architecture. In comparison to amphiphilic block copolymer micelles, PCMs are far less quantitatively understood at a molecular level. As several underlying features complicate the thermodynamic framework of PCMs. For example, the ionic core consists of two distinct polyelectrolytes that, under stoichiometrically equivalent charge matched conditions, form intrinsic ion pairs that act as physical cross-links between polycation and polyanion repeat units. These pairings can be disrupted via the addition of salt or by heating, making PCMs highly responsive to changes in the local environment. The low interfacial tension and water solubility of polyelectrolyte chains in complex coacervates means water is present throughout both the core and corona, further complicating efforts to understand the fundamental physics of these nanoparticles, owing to their intrinsically multicomponent nature.

In this perspective, we discuss a collection of recent research articles that shed new light on design strategies for dilution solutions of PCMs using integrated measurement, analysis, and prediction from experimental and computational tools. Special attention is given to the development of (i) scaling relationships governing size, shape, and morphological transitions of PCMs, and (ii) micellization dynamics in PCM formation/growth, chain exchange, and disassembly pathways. We also provide direct examples of extending polyelectrolyte structure–property principles to impart favorable physicochemical attributes for delivery applications and discuss future directions. Unless specifically mentioned, the studies reviewed here use fully ionized strong polyelectrolytes at stoichiometric charge ratios. Recent reviews on PCM use in gene therapy and molecular interactions in polyelectrolyte complexation can provide further insight into the broad state of this field. These selected works provide blueprints for advancing our fundamental understanding of this important class of self-assembled materials.

2. PCM STRUCTURE–PROPERTY RELATIONSHIPS

Controlling the size and structure of PCMs is vital to their success as therapeutic delivery vehicles. Nanoparticles that are smaller than \( \sim 10 \text{ nm} \) may be removed from the bloodstream by the kidneys, whereas nanoparticles above 200 nm are prone to nonspecific accumulation in the spleen and liver. In addition to avoiding renal clearance, nanoparticle design can have profound impacts on nanoparticle biodistribution and cellular uptake. Recently, Ridolfo et al. explored morphology effects in biological settings by comparing the cellular uptake of amphiphilic spherical, worm-like, vesicular, and tubular nanoparticles. They found that higher aspect ratio nanoparticles such as worms and tubes performed better than spheres and vesicles because higher aspect ratio nanoparticles diffused faster relative to low aspect ratio nanoparticles. These observations should apply to amphiphilic micelles and PCMs equally well, as these observations do not rely on the assembly mechanism. For these reasons, precise control of PCM size and morphology is a key component to developing efficient PCM encapsulants. This section covers structure–property relationships of PCMs with a summary in Table 1.

Morphology of PCMs. The length of each polymer block (A, B, or C in Figure 1) can dictate PCM morphology. Roughly, if the length of the neutral block is larger than that of the charged block, i.e., the neutral/charged length ratio \((N/\text{C}) > 1\), self-assembly results in spheroidal micelles for both \((AB + AC)\) and \((AB + C)\) systems. When \(N/\text{C} < 1\) other morphologies can be formed. For most \((AB + C)\) systems, these assemblies are aggregates and complexes similar to bulk assemblies, as the small fraction of neutral polymer does not force microphase separation. However, for \((AB + AC)\) systems, interesting morphologies can be formed in the \(N/\text{C} < 1\) regime. For example, the Kataoka group observed polyelectrolyte complex vesicles when \(N/\text{C} \sim 0.5\) compared to spheres for the same system with \(N/\text{C} \sim 2\). Cylindrical and planar assemblies have also been observed when \(N/\text{C} < 1\) with a very low poly(ethylene oxide) (PEO, also referred to as poly(ethylene glycol) or PEG) weight fraction. PCM theory predicts that for \(N/\text{C} \ll 1\) morphology scales with degree of ionization \((f)\). As \(f\) increases, morphology changes to lamellae, cylinders, crew-cut spherical micelles (corona thickness \(\ll\) core size), and finally star-like...
spherical micelles (corona thickness $\gg$ core size). At very low $f$, where the free energy gain of complexation is on the order of thermal energy, micellization does not occur and a solution of unimers occurs. Likewise, when charged blocks are very short, experimental results show minimal complexation.30,37 Other unimers occurs. Likewise, when charged blocks are very short, thermal energy, micellization does not occur and a solution of where the free energy gain of complexation is on the order of experimental results show minimal complexation.30,37

Table 1. Molecular Architecture Controls Polyelectrolyte Complex Micelle (PCM) Properties

| property                  | parameter       | notes             | references |
|---------------------------|-----------------|-------------------|------------|
| core radius               | neutral block length ($N_0$) | inverse correlation | 36, 59     |
| core radius               | charged block length ($N_C$) | direct correlation | 36, 37, 56, 59, | 77−79 |
| core radius               | homopolymer length ($N_h$) | independent       | 30, 37, 42, 56, 59 |
| corona thickness          | neutral block length ($N_0$) | direct correlation | 36, 59−61   |
| aggregation number        | neutral block length ($N_0$) | inverse correlation | 30, 59−61  |
| aggregation number        | charged block length ($N_C$) | direct correlation | 36, 59, 64  |
| aggregation number        | strength of charge | direct correlation | 50         |
| polydispersity            | strength of charge | direct correlation | 37, 50     |
| morphology                | N/C $> 1$       | spheroids         | 30−32      |
| morphology                | N/C $< 1$       | aggregates, vesicles, or lamellae | 30, 33−35, 40, 41 |
| morphology                | polyelectrolyte architecture | vesicles, cylinders, and more | 38, 39 |
| morphology                | DNA hybridization | ssDNA $\approx$ spheres | 37, 56 |
| morphology                | dsDNA $\approx$ worm-like cylinders | 37, 56 |
| stability                 | polymer length  | direct correlation | 66         |
| stability                 | charge density  | direct correlation | 67, 68     |
| stability                 | strength of charge | direct correlation | 50         |
| stability                 | cross-linking   | direct correlation | 73−76      |

The following experimental trends are from both experimental and theoretical publications on PCMs.4 Under lengths of $\sim$5000. *Specific cases.

The length of each polymer block in a spheroidal micelle can influence its structural properties including size, aggregation number, and stability. In (AB + C) systems, the PCM core radius ($R_{core}$) is directly proportional to the length of the charged block in the block copolymer ($N_C$), which largely independent of the length of the homopolymer ($N_h$), at least below a large critical length around $N \approx 5000$. The size of the neutral block ($N_0$), which forms the corona, has to have a minor effect on the size of the core, but noticeably drives the thickness of the corona ($H$) and therefore the hydrodynamic radius ($R_h$), overall size of the micelle, which is a crucial parameter for controlling biodistribution.60,61 Conversely, the aggregation number ($P$), or number of chains in a given micelle, is shown to decrease as the neutral block size increases for a (AB + C) system.59−61 To quantify these physical trends, the group developed experimental scaling laws for $R_{core}$, $R_h$, $H$, and $P$ using PCMs containing PEO-block-pLys (PEO-b-pLys) paired with ssDNA or pGlu.58 These scaling laws are shown as black lines in Figure 2 overlaid with accumulated published (AB + C) PCM data representing a variety of synthetic and biological polymers. The data were normalized using the scaling laws for two polymer lengths and plotted against the third length variable. This normalization collapsed all the data to a single trend and is compared to the corresponding scaling law in this figure. $N_0$ was found to have no noticeable effect on any physical parameter, which is convenient for creating therapeutic delivery systems where the C component is often a therapeutic drug or biomolecule. Polyelectrolyte length is generally reported as degree of polymerization, as it is considered here for physical scaling,
but contour length or physical size are likely a slightly more accurate factor.

Theoretical work on PCMs is more precise and predicts similar physical property scaling relationships for two geometrical extremes of PCM structure.36,62,63 For PCMs with fully ionized chains in a "good" aqueous solvent, predictions for $R_{\text{core}}$, $P$, and $H$ at the star-like limit, where ($R_{\text{core}} \ll H$), are shown in eqs 1–3.

$$R_{\text{core}} \propto N_B^{0.6}$$

$$H \propto N_A^{0.6}N_B^{0.2}$$

$$P \propto N_B^{0.8}$$

Scaling theory for the same conditions, but at the crew-cut limit where ($R_{\text{core}} \gg H$), are shown in eqs 4–6.

$$R_{\text{core}} \propto N_A^{-0.5}N_B^{1}$$

$$H \propto N_A^{0.8}$$

$$P \propto N_A^{-1.6}N_B^{2}$$

The PCMs in Figure 2 consist of fully ionized chains in good solvents and are between the star-like and crew-cut regimes, which is the case for the majority of experimental work. Considering the intermediate regime of these PCMs, the experimental scaling laws39 shown in Figure 2 are consistent with predictions for the two structural limits. Further theoretical predications show dependence on solvent quality, salt concentration, and degree of ionization36,62–65 but are out of the scope of this structural review. Understanding how PCM structural properties are controlled by polymer structure can accelerate the design process for tailored carriers.

**Stability of PCMs.** One aspect of PCM stability is measured by its resistance to nanoparticle degradation in the face of increasing ionic strength in solution. Adding excess counterions from salt competes with ion pairing between polymers to disrupt complexation. In bulk systems (B + C), increasing the length of either charged polymer increases the stability,54 which can be easily tested with optical microscopy. This is more difficult to study for nanoparticles that are smaller than the diffraction-limited resolution of optical microscopes, but light scattering and small-angle scattering techniques have been used to show a similar effect for PCMs.42,66 Likewise, increasing charge density increases complex stability67 and can drive micellization in PEC systems with charged biomolecules.68

PCMs are often discussed as two charged, flexible chains coming together in an entropically favorable process; however, the molecular details of each charged group also play a role in the structure and stability of complex formation.69 The Choi group used a functionalizable PEO-b-PAGE for a direct comparison of PCMs comprising charged ammonium ($pK_a = 11$), guanidinium ($pK_a = 14$), carboxylate ($pK_a = 4$), and sulfonate ($pK_a = 1$) groups using thiol–ene click chemistry to attach each of the desired side groups onto otherwise identical polymers. The neutron scattering results revealed an increase in core radius and aggregation number as ion pairing interactions become stronger.50 Our group compared two cationic charged monomers in comparable polymer structures: lysine (primary amine, $pK_a = 10$) and vinylbenzyltrimethylammonium (VBTMA, consisting of a permanently charged ammonium) for complexing DNA at various lengths.37 Despite the permanent charge and additional hydrophobicity imparted by the aromatic moiety, which have been previously shown to strengthen certain PEC systems,70,71 PVBTMA complexed less strongly with DNA, attributed to the steric hindrance in ion

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**Figure 2.** Aggregated data from published (AB + C) polyelectrolyte complex micelle (PCM) experimental studies using strong polyelectrolytes at stoichiometric charge ratios, overlaid with experimental scaling laws shown as black lines. The data were normalized using scaling laws for two block lengths and plotted against the third block length, collapsing to show scaling for the block length of interest. The available literature provides aggregated data for core size (A–C), hydrodynamic size (D–E), and aggregation number (F). The data represents PCMs from six publications30,37,42,56,59,60 using numerous synthetic and biological polymers and the scaling laws are experimental,59 consistent with theoretical predictions36 for PCMs between the star-like and crew-cut regimes. Adapted from Marras et al.59 Copyright 2021 American Chemical Society.
that precise lengths of neutral and charged blocks can be prepared with low dispersity in the molar mass distribution. Depending on the block lengths and pairing of the PEO-b-PVBTMA polycation with PAA, PEO-b-PSS, or PSS polyanion, the assemblies that form can resemble spherical, core/shell PCMs, or polydisperse colloidal aggregates. Both classes of nanostructures will be discussed in detail below, along with open-ended questions that these results raise for the physical chemistry community.

**PCM Formation Kinetics.** Following established stopped-flow protocols from amphiphilic block copolymer literature, the ultrafast formation of PCMs can be monitored in situ using time-resolved small-angle X-ray scattering (TR-SAXS) with millisecond temporal resolution (Figure 3B). Here, solutions of oppositely charged polymers are loaded into separate syringes, pumped into a turbulent mixer, and dispensed into a capillary cell without further flow for scattering measurements. This technique has provided new physical insights into ionic nanomaterial behavior, including complex coacervate coalescence of poly(allylamine hydrochloride) and PAA as a function of added NaCl salt. However, it was only very recently that the initial complexation of block polyelectrolytes has been reported. The chemical and electrostatic nature of the polyelectrolyte pairing appears to greatly influence the kinetic pathway of micellization, demonstrating the importance of mindful polymer selection in constructing PCM nanocarriers. Two independent cases that illustrate completely different pathways are shown in Figure 4.

In the first case, Wu and co-workers investigated the spatiotemporal formation kinetics of PEO-b-PVBTMA with PAA. Using a stopped-flow apparatus with high-throughput data collection capabilities at the Stanford Synchrotron Radiation Lighthouse (SSRL, SLAC National Accelerator Laboratory), they directly observed the assembly kinetics and SAXS profiles of PEO-b-PVBTMA/PAA PCMs via TR-SAXS from 100 ms to 5 s, which exhibited spherical particles (\( q^0 \) power law dependence of intensity for \( q < 0.01 \) Å\(^{-1} \)) that grew in size over time (Figure 4A). The structural evolution of PCMs was evaluated by determining the apparent Guinier radius of gyration \( (R_g) \), which showed incremental micelle growth from \( R_g \approx 10 \) to \( R_g \approx 12 \) nm over 5 s by gradual insertion of either unimer chains or ion-paired clusters. For the second case, Amann and co-workers examined PCMs comprising PEO-b-PVBTMA and PSS. Using a SFM-400 stopped-flow apparatus at the European Synchrotron Radiation Facility (ESRF), the researchers reported the TR-SAXS formation of metastable aggregates for internal charge neutralization at 3 ms, preceding rearrangement and pinch-off into small micellar particles over the course of 30 s (Figure 4B). The equilibration data was described by these relaxation processes as a function of the degree of polymerization \( (N) \) of PVBTMA from \( N = 6 \) to \( N = 19 \), where rearrangement of unimer chains or ion-paired clusters becomes increasingly unfavorable as block length increased.

In attempts to account for the differences in kinetic pathways, we have shown that the homopolymers PVBTMA + PAA form liquid-like coacervates, whereas PVBTMA + PSS form solid-like complexes. This observation leads us to speculate that the formation kinetics may be strongly dependent on the chemical nature of the polymer constituents, though further work needs to be done to test this hypothesis of whether the complex cores resemble the nature of macroscopic complexes. In addition, block length of the block polyelectrolytes may also affect the rate
at which PCMs either grow incrementally or break apart from larger colloidal structures upon complexation. In a previous report, Wu et al. showed that PEO-$b$-PVBTMA/PAA and PEO-$b$-PSS at $N \approx 50$ for the charged blocks form nonequilibrium complexes, far from well-defined spheres by fitting the SAXS data. Investigating this system that matches the neutral and charged blocks lengths to PEO-$b$-PVBTMA/PAA as a direct comparison to the results of Amann et al. is the subject of ongoing work. Altogether, these examples illustrate only two of many possible formation pathways that lead to charge-driven micellization. Expanding TR-SAXS to study more PCM systems and varying parameters like polyelectrolyte selection, block lengths, and molecular architecture can help move PCM design toward more efficacious and predictive encapsulation of cargo.
PCM Chain Exchange. Understanding chain exchange in dilute micelle solutions is crucial toward their development as efficient delivery vehicles, as the rate and method by which sequestered therapeutic molecules are exposed to the surrounding environment will control their efficacy. In general, chain exchange between equilibrium polymeric micelles proceeds via two primary mechanisms: chain expulsion/insertion and micelle fusion/fission. While chain exchange processes have been probed with amphiphilic polymer assemblies, to the extent of our knowledge very few experimental studies have examined chain exchange for PCMs. Fluorescent imaging is one accessible approach to potentially examine the underlying exchange mechanism of PCMs. Nolles and co-workers have utilized fluorescently labeled proteins in PCMs to probe formation kinetics and exchange dynamics. Synchrotron scattering is another potential strategy. We have previously shown how interparticle effects emerge in the form of a structure factor in SAXS profiles for concentrated micelle solutions and thereby maintain PCM stability over time, but the molecular details of unimer chain exchange or fusion/fission cannot be unveiled with these experiments, as they lack molecular contrast. Unfortunately, scattering methods with appropriate contrast such as SANS require long time scales and deuterated/hydrogenated systems, limiting their applicability and accessibility. However, advances in molecular dynamic simulations have offered new insights to potential mechanisms of exchange in which electrostatics interplay with other competing noncovalent interactions.

Bos et al. performed coarse-grain dynamics simulations on a model diblock-homopolymer PCM system and outlined physical characteristics influencing the mode of chain exchange and PCM stability. In these Langevin dynamics simulations, nonelectrostatic interactions (i.e., hydrogen bonding or hydrophobicity) tended to disfavor the chain expulsion mechanism, as these additional interactions generate an enthalpic cost which counteracts the entropic gain that drives small neutral complex cluster expulsion from PCMs (Figure 5A). Interestingly, these effects depended on the nonelectrostatic interactions (represented by the Lennard-Jones potential $\epsilon_{ij}$) being intermolecular and changed when the interactions were modified to be solely intramolecular. In the case where one polyelectrolyte displayed significant nonelectrostatic interactions with itself but not the oppositely charged polyelectrolyte, chain expulsion and fusion/fission events increased, suggesting that the PCMs became less stable. Meanwhile, macromolecular design parameters such as block length have divergent effects on the two mechanisms as $\epsilon_{ij}$ was increased, shown in Figure 5B. Chain expulsion was sensitive to the relative length of the charged blocks in the system, displaying a distinct increase for matching lengths relative to unmatched chain lengths. Fusion/fission, on the other hand, appeared insensitive to the ratio of the block lengths, but instead depended on the total length of the homopolymer. Taken together, simulation and experiment provide a strong case for the careful consideration of nonelectrostatic interactions between polyelectrolytes and polyelectrolyte length.

Bos, Timmerman, and Sprakel more recently demonstrated the exchange dynamics of PCMs using Förster resonance energy transfer (FRET). In this work, poly(ethylene oxide)-b-poly(trimethylammonioethyl methacrylate chloride) (PEO-b-PTMAEMA) and fluorescently labeled poly(3-sulfopropyl methacrylate) (PSPMA) were used as the model system. An analytical model relating the FRET efficiency between fluorophores and the exchange rates of polyelectrolyte chains was developed, in which the broad distribution of detected exchange rate was correlated to changes in salt concentration, polymer length, and micelle concentration. To the extent of our knowledge, this is the first experimental demonstration that PCM equilibration occurs predominately through expulsion/insertion pathways instead of fusion/fission.

For PCMs containing nucleic acids, an important practical consequence of dynamic chain exchange between micellar assemblies is the potential displacement of nucleic acids over time upon immersion in different biological settings. Because of the complicated delivery pathways involved with overcoming various biological barriers, molecular engineering approaches to boost stability in nanocarriers are nontrivial. However, simplified fundamental experiments have the potential to reveal structure–property relationships for PCM stability, answering questions that bring us closer to this goal. For example, can foreign polyelectrolytes with a strong tendency of association into PCM hosts result in mixed micelles? If so, what features are consequential for this feature of macromolecular exchange? To the extent of our knowledge, there are only a handful of published works that have examined such questions. Dautzenberg et al. conducted polyanion exchange reactions involving PCMs containing model oligophosphates and competing higher MW polymers (PSS and DNA) at physiological salt conditions. Another study by Harada and Kataoka exploited polymer architecture to show how diblock polyelectrolytes displace homopolyelectrolytes in PCMs formed with an oppositely charged diblock polyelectrolyte, suggesting increased association in (AB + AC) PCM systems compared to (AB + C) systems. These examples provide insight on how PCM chain exchange can arise from molecular recognition based on polyelectrolyte compatibility and dynamics, an area in which advances in noninvasive characterization techniques such as small-angle neutron scattering, fluorescence microscopy, and liquid-phase transmission electron microscopy are well-equipped to provide key insights in our understanding of polyelectrolyte complex micellization and chain exchange.

PCM Disassembly During Sudden Environmental Changes. Depending on the intended application, PCMs may encounter gradients in ionic strength, pH, or temperature that may disrupt their structure. It is important to understand what implications dynamic environments may have on PCM stability or disassembly. One straightforward way to evaluate the dissociation of PCMs is through stopped-flow light scattering (Figure 3B), where PCMs can be monitored in situ while a sudden environmental step change, such as a temperature or salt, is introduced.

In this setup, the time-dependent evolution of the scattering intensity is directly proportional to the mean aggregation number $\langle P(t) \rangle$ and the salt-induced temporal dissociation can be fitted by a phenomenological Avrami-type compressed exponential function of the form in eq 7, containing the relaxation time ($\tau$) and the exponential ($\beta$) related to nucleation/growth:

$$P(t) \approx e^{-(t/\tau)^\beta}$$

Wu et al. examined the disassembly kinetics of PEO-b-PVBTMA/PAAC PCMs with stopped-flow light scattering at different temperature and salt conditions. Table 2 shows the fitting results at (i) 20, 37, and 57 °C, with a constant salt jump of 500 mM NaCl, and at (ii) 300, 400, 500, and 600 mM NaCl, with a constant temperature of 20 °C. As temperature was increased, the dissociation process accelerated as $\tau$ decreased.
from 61.5 to 39.2 min, with $\beta$ set to 2 assuming second-order kinetics associated with a fission/fusion mechanism of micelle fragmentation and separation. Analogously, by varying the salt-jump concentration and allowing both $\tau$ and $\beta$ to change, higher salt resulted in faster relaxation kinetics, though disassembly was not immediate after the salt-jump. In addition, the $\beta$ changed from approximately 1 to 2, showing that neither single-chain expulsion/insertion nor the fission–fusion mechanism were the full explanation for PCM disassembly kinetics. Overall, this technique provides a way for researchers to quantitatively assess the robustness of PCMs under changing solution environments.

### 4. APPLICATIONS AND OUTLOOK

Fundamental studies on physical scaling and kinetics continue to pique scientific interest and deliver a molecular-level understanding of PCMs with an ultimate goal, from our perspective, of creating tunable polymeric nanoparticles for biomolecule delivery. When model polymers are substituted for therapeutic biomolecules and PCMs are put into complex environments new problems arise and the research adapts. This field has a strong reciprocal connection between fundamental research and application-based science which is apparent in the numerous advances propelling the potential of using PCMs for biomolecule delivery.

**Biomolecule Delivery using PCMs.** Due to their controllable nanoscale size and morphology, distinctive capability of partitioning hydrophilic cargo, and dynamic responsivity to environmental changes and stimuli, PCMs are well-suited as delivery vehicles for nanoscopic cargo. Nonviral delivery of therapeutics is a critical challenge for nanomedicine and has been evolving for decades.105–108 PCMs are unique among therapeutic nanoparticles in that they are assembled from only hydrophilic materials and thus, are highly hydrated. Compared to hydrophobically driven assemblies, PCMs do not suffer from certain burdensome limitations on biodistribution such as accumulation in the liver.109 They also have a unique ability to sequester the inherently anionic nucleic acids in the core of the micelle, protecting the cargo from harsh environments and enzymatic threats. These studies include cellular delivery and animal models, driving PCM research toward real world applications in gene therapy and immunization.

Proteins are incorporated in polyelectrolyte complexes, largely for applications in bulk materials,119 but also in PCMs for delivery. Proteins have both anionic and cationic amino acids on their surface, making sequestration in PCMs less straightforward compared to nucleic acids. Strongly charged proteins can form polyelectrolyte complexes rather simply, while proteins that are closer to net-neutral at physiological pH must be converted to a stronger charge. Strategies for charge conversion include adding more charged amino acids to native proteins,120 working at a pH beyond the protein’s isoelectric point or by modifying the actual charged groups on the protein. One method uses citraconic acid to convert primary amines to carboxylic groups on the protein surface, making the protein’s net charge more negative to enable complexation. These groups can be converted back to their original cationic state in acidic conditions like the late endosomal environment. The Kataoka group has demonstrated this strategy by changing the charge of antibodies in order to assemble PCMs with diblock polyelectrolytes including PEO-b-pLys.121–123 Under acidic conditions, natural antibody charge is restored, disrupting micellization and releasing the antibodies while restoring their biological activity. This strategy was successfully implemented for delivery into cancer cells122,123 and to the brain124 using glucose on the PCM surface to cross the blood–brain barrier with glucose transporter proteins. Strengthening association through nonionic methods can also improve protein–polyelectrolyte complex stability, such as adding hydrophobic spacers to polyelectrolytes.124 Complex macromolecules with multiple components like the ribonucleoprotein complex used in CRISPR-Cas9 genome editing can also be sequestered and delivered in PCMs,75,125 expanding their versatility.

**Major Challenges and Opportunities.** We have covered advancements in tailored PCMs with a focus physical and dynamic characteristics and recent developments toward therapeutics. This success to date, however, is just a fraction of the enormous potential PCMs have in advancing nanomedicine through enhanced biodistribution, targeting, and controlled release for applications including gene therapy, gene manipulation, and protein- or peptide-based drug delivery. The future of PCMs for nanomedicine will build on these efforts and those in other areas of medical research and could have an important role in the accelerating need for nanotherapeutics. The COVID-19 pandemic has accentuated the growing importance of versatile biomolecule carriers. In addition to vaccines, nanoparticle delivery has promising applications in therapies using gene silencing, monoclonal antibody therapy, and small molecule immunootherapy. PCM research has demonstrated the capability to accomplish these tasks using completely hydrophilic components for improved distribution, tailored particle size, shape, stability, release, and custom surface modifications, but many hurdles remain.

PCMs have been shown to be effective in cellular delivery, but the complete pathways are largely unknown and require further investigation. Advancing cell targeting will vastly improve the effectiveness of therapeutic PCMs. Attaching folate to the corona can mediate delivery to cells that overexpress the folate

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**Table 2. Relaxation Rate and Exponential Fits of Micelle Dissociation at Increasing Temperature and Salt Concentrations**

| temp. (°C) | [salt] (mM) | $r$ (min) | $\beta$ |
|-----------|-------------|-----------|--------|
| 20        | 500         | 61.5      | 2.00   |
| 37        | 500         | 52.2      | 2.00   |
| 57        | 500         | 39.2      | 2.00   |
| 20        | 300         | 51.9      | 0.82   |
| 20        | 400         | 35.2      | 1.43   |
| 20        | 500         | 27.4      | 2.03   |
| 20        | 600         | 10.9      | 1.94   |
receptor,126,127 like breast cancer. Sequence-defined peptides have been demonstrated for targeting to inflamed vascular endothelial cells.112 Likewise, RGD peptides can promote cell adhesion,128,129 showing biodistribution and targeting cancer cells that upregulate integrins.130 PCM behavior and stability within the weakly acidic environment of an endosome should also be further characterized. RNA, in particular, is sensitive to acidic conditions and must continue to be protected during this stage. Ultimately, the cargo must escape the endosome and be released from the PCM into the cytoplasm, posing a major concern for PCMs. However, preliminary work, using endosomolytic peptides131 and membrane disruption through deprotecting cations,132,133 shows promising results for programming endosomal escape and nucleic acid release.

Selection of physical and chemical polymer attributes is the central tool among the vast PCM design space we have discussed here. These efforts are crucial when designing a delivery system that needs to protect and release cargo in complex environments. Many studies have shown that increasing charge density or polyelectrolyte length will improve salt resistance, indicating a stronger complex. The McCormick group has shown that longer polyelectrolytes increase PCM effectiveness when using siRNA for gene knockdown and silencing applications but with a time delay due to increased binding constants.66 Additionally, lowering polyelectrolyte binding strength by reducing charge density improves siRNA release but makes PCMs more susceptible to enzymatic degradation, ultimately decreasing cell transfection efficiency.67 Furthermore, it is well documented that increasing polycation charge density or molecular weight increases cytotoxicity.134–136 Achieving an acceptable balance between release kinetics, transfection efficiency, and managing cytotoxic effects stresses the importance of polymer selection when designing therapeutic micelles.

We have reviewed the ways polymer selection in core-forming blocks affects PCM size, stability, and efficacy, but polymer choice for neutral blocks is not frequently studied, likely because the current standards work well. Neutral hydrophilic polymers that form nanoparticle coronas are most commonly PEO, which is easily soluble in aqueous solutions, commercially accessible, and shows little concern in vivo, however, better options are seldom explored. Recently zwitterionic polymers have been incorporated into PCMs as net-neutral blocks because of the excellent antiprotein resistance, hydrated lubrication properties, and high biocompatibility.65,137 Protein-resistant corona materials give a route to increased nanoparticle stealth as protein adsorption to the corona is a significant mechanism contributing to nanoparticle expulsion from the bloodstream.138 This field can benefit from a greater understanding of zwitterionic corona behavior and its effect on biomolecule delivery.

The platform for delivering nucleic acids, a variety of proteins, and some small molecule drugs139 using PCMs is already established and unique strategies for selective delivery are continuously being unveiled. Understanding the factors that influence PCM physical properties, stability, and disassembly will be crucial when designing for delivery. Tailored PCMs that protect, deliver, and release therapeutic cargo with control over transport and targeting can enhance precision medicine, driven by advances in structural design.

Author Contributions
The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Notes
The authors declare no competing financial interest.

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