Formation and enhancement of negative thermal quenching in emission of KGdF₄:Eu³⁺, Yb³⁺@GQDs†

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In order to obtain a red emitting phosphor with good luminescence thermal stability, a series of KGdF₄:Eu³⁺, Yb³⁺@GQD (GQD: Cl-containing graphene quantum dots) red emitting phosphors have been synthesized by the co-precipitation method, and their luminescence thermal properties have also been studied in detail. It is intriguing that the negative thermal quenching (NTQ) effect is induced by the double doping of Yb³⁺, and the effect is further enhanced by GQD coating. The strongest integrated PL intensities of the optimal double doped sample and the optimal GQD-coated sample are at 130 and 170 °C, in which the corresponding integrated PL intensities are presented as 117.7 and 156.5% of the initial value at 30 °C, respectively. The NTQ effect makes the optimal GQD coated sample have good luminescent thermal stability, so it can be applied for high-power WLEDs. A mechanism of energy conversion from heat to light is discussed and suggested for the effect.

1. Introduction

White light emitting diodes (WLEDs) have gradually become the mainstream lighting source due to their characteristics of high efficiency, low energy consumption, low heat dissipation, long life, good light quality, energy saving and environmental protection.1–3 The traditional method to obtain white light is to coat the yellow emitting phosphor on the blue LED chip, and realize the white light emission by mixture of the blue light of the chip and the yellow light of the phosphor. But there are some serious problems in this system, such as thermal quenching, poor color reproducibility, high color temperature (CCT > 5000 K), and low color rendering index (CRI < 80) due to the lack of red emitting components, which have a great impact on practical application.4–6 Another method to obtain white light is to coat three primary color (blue, yellow and red color) phosphors on near-ultraviolet (near-UV) light LED chip (370–410 nm) and achieve white light emission by mixture of the three primary color lights. The phosphors are the focus of current research on luminescent materials. However, the red emitting phosphors obtained in this way are mainly Y₂O₃:S:Eu³⁺. Their luminescent efficiency is far lower than that of blue and green phosphors and their chemical properties are unstable. It decomposes easily at WLEDs working temperatures to release toxic hydrogen sulfide and reduce the service life of WLEDs.7 Therefore, it is urgent to develop a red emitting phosphor with high luminescent thermal stability, which is suitable for WLEDs excited by near-UV light (370–410 nm).

Eu³⁺ ion is generally recognized as a good activator for red phosphor. Under near-UV (370–410 nm) excitation, 5D₀ → 7F₂ (j = 0, 1, 2, 3, 4) transitions emit red light, so that many researchers are trying to develop Eu³⁺ as activator to develop red emitting phosphor. But because of the shielding effect of 5s and 5p electrons in the outer shell, Eu³⁺ has weak luminescent intensity and small absorption cross section. So it is necessary to find an appropriate sensitizer to improve the luminescence intensity.8–14 Eu³⁺ and Yb³⁺ often co-doped in various kinds of matrix, for example: Lu₂O₃,15 CaF₂,16,17,18 Gd₂O₃,19 KGdF₄ (ref. 16) and NaYF₄,20 etc., and emission of Eu³⁺ is enhanced by absorbing the infrared light though Yb³⁺. But, as best as we know, that under near-UV (370–410 nm) excitation, the emission enhancement of Eu³⁺ by co-doping of Yb³⁺ has been rarely reported.

In addition to luminescent intensity, thermal quenching of phosphors is also an important factor affecting the luminescent performance, and the Eu³⁺ doped red emitting phosphors are no exception. The luminescent intensity of Eu³⁺ will decrease with the increase of temperature in most matrices, such as Ba₃Bi(PO₄)₃:Eu³⁺,18 Sr₂Al₂O₆:Eu³⁺,19 K₄CaSiO₄:Eu³⁺,20 Ca₂LaTaO₇:Eu³⁺,21 Ba₃ZrNb₂O₁₀:Eu³⁺,22 etc. However, the red emitting phosphors doped with Eu³⁺ do not completely decay, and their luminescent intensity is still more stable than that of Eu³⁺ doped phosphors. So it is urgent to develop a red emitting phosphor with high luminescent thermal stability, which is suitable for WLEDs excited by near-UV light (370–410 nm).

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KGdF₄ is a good matrix for doping of Eu³⁺ ion. Eu³⁺:Sm³⁺ phosphor was induced to have the NTQ effect. Eu³⁺ or Yb³⁺ is within 15%. Based on the Vegard Law, various down-conversion and up-conversion materials. Stability and high refractive index. So, it is widely used as matrix for matrix due to its high ionicity, low phonon energy, good optical properties. Other, which facilitates co-doping of Eu³⁺ and Yb³⁺. Therefore, ion radius does not exceed 15%, they can easily replace each other, which facilitates co-doping of Eu³⁺ and Yb³⁺. Therefore, KGdF₄ is a good matrix for doping of Eu³⁺ ion.

So, as a part of our system research, in this paper, based on KGdF₄:Eu³⁺, Yb³⁺ obtained via a co-precipitation method, GQDs is coated on the surface of KGdF₄:Eu³⁺, Yb³⁺ to form KGdF₄:Eu³⁺, Yb³⁺@GQDs hybrid composite luminescent materials. Some novel effects induced by co-doping of Yb³⁺ and coating of GQDs have been observed, which the NTQ effect was formed after co-doping of Yb³⁺ and it was further obviously enhanced by coating of GQDs.

2. Experimental procedure and methods

Experimental section is described in ESI.† Experimental section includes reagents, apparatus, and synthesis of the samples. Properties of GQDs are also described in (Table 1S, and Fig. 1S†).

3. Results and discussion

3.1 Structure, morphology and composition

The ICP elemental analysis data of samples (i–iii) are listed in Table 1, which the corresponding molecular formulae of the samples (i–iii) are KGF:0.20Eu³⁺, KGF:0.20Eu³⁺, 0.002Yb³⁺, KGF:0.20Eu³⁺, 0.002Yb³⁺@GQDs₂ mg mol⁻¹. Fig. 1 shows the XRD patterns of the samples (i–iii). It can be seen from Fig. 1 and Table 2 that: first, doping of Eu³⁺, Yb³⁺ and coating of GQDs do not change the structure of the samples. Second, the crystallinity of the samples is relatively high, and there is no any other impurity peaks. Finally, the crystal structures of the samples are similar to the standard pattern of cubic NaGdF₄ (PDF # 27-0697), but the position of the all peaks are shifted to a small angle, due to the ionic radius of K⁺ being larger than that of Na⁺.

In order to confirm the phase of the samples (i–iii), we not only synthesized NaGdF₄:0.20Eu³⁺ by the same method (Fig. 7S†), but also compared with the XRD patterns of KGdF₄:Ln³⁺ (Ln = Eu, Ce, Sm, Tb) reported by other authors. Fig. 7S† shows that XRD pattern of the as-synthesized NaGdF₄:0.20Eu³⁺ is in agreement with those of hexagonal NaGdF₄ (PDF # 27-0699) and hexagonal NaEuF₄ (PDF # 49-1897), but it does not match to that of cubic NaGdF₄ (PDF # 27-0697). Some authors illustrate that the XRD patterns of KGdF₄:Ln³⁺ (Ln = Eu, Ce, Sm, Tb)³⁸ are close to those of hexagonal YGdF₄ (PDF # 27-0697), but they are not similar to that of orthorhombic KGdF₄ (PDF # 33-1007). It is interesting that the XRD patterns of the samples (i–iii) are the same as those of KGdF₄:Ln³⁺ (Ln = Eu, Ce, Sm, Tb),³⁸ indicating that they are phase of KGdF₄.

Fig. 2 shows the SEM image, EDS, XPS and FTIR spectra of sample (iii), KGF: 0.20Eu³⁺, 0.002Yb³⁺@GQDs₂ mg mol⁻¹. Fig. 2a (SEM) shows that the sample is cluster of irregular particles with a size of about 5–10 μm, and many small particles are adsorbed on the cluster surface. Fig. 2b (EDS) shows that the sample contains elements of K, Gd, Eu, Yb, F and C, indicating that GQDs has been successfully coated on surface of the sample. XPS results (Fig. 2c) show that the sample is composed with K, Ti, Mn, F, C and O, which C and O come from GQDs, adsorbed CO₂ and H₂O, respectively. FTIR spectra (Fig. 2d) show that sample (iii) and GQDs have the same characteristic FTIR absorption peak (Fig. 6S†). Here,

| No. | K/% | Gd/% | Eu/% | Yb/% | Atomic ratios of K : Gd : Eu : Yb | Calculated molecular formulae |
|-----|-----|------|------|------|---------------------------------|--------------------------------|
| (i) | 14.327 | 46.455 | 11.237 | 0.000 | 0.996 : 0.803 : 0.201 : 0.0000 | KGF:0.20Eu³⁺ |
| (ii) | 14.311 | 46.172 | 11.068 | 0.146 | 0.995 : 0.798 : 0.198 : 0.0023 | KGF:0.20Eu³⁺, 0.002Yb³⁺ |
| (iii) | 14.354 | 46.149 | 10.957 | 0.134 | 0.998 : 0.798 : 0.196 : 0.0021 | KGF:0.20Eu³⁺, 0.002Yb³⁺@GQDs₂ mg mol⁻¹ |

Table 1. ICP element analysis results of samples (i–iii) Encyclopedia of materials: properties, synthesis, processing, and applications. 2011. © 2011 The Author(s). Published by the Royal Society of Chemistry RSC Advances.
the characteristic sharp peak at 1639 and 3450 cm$^{-1}$ can be attributed to C=C vibration of GQDs, and O–H vibration of H$_2$O, confirming that surface of the sample has GQDs. In conclusion, results of EDS, FTIR and XPS all illustrate that GQDs has been coated on the surface of the sample (iii).

3.2 Luminescence properties

Fig. 3 shows luminescent properties of the three samples at room temperature. As shown in Fig. 3a, the strongest excitation peaks of the samples are at about 393 nm. Co-doping of Yb$^{3+}$ and coating of GQDs significantly enhances and widens the peaks, and the intensity order of the peaks is (iii) > (ii) > (i). Fig. 3b shows PL spectra of the samples, which the emission intensities are also significantly enhanced by Yb$^{3+}$ co-doping and GQDs coating, but the position of the main emission peaks has not changed: (a) emission peak intensity of the sample (ii) is 1.40 times that of the sample (i); (b) emission peak intensity of sample (iii) is approximately 1.20 times that of the sample (i) and approximately 1.68 times that of the sample (i). (c) It can be seen from Fig. 4Sb,† GQDs should have an emission peak at about 460 nm under excitation of 393 nm. However, in Fig. 3b, no emission peak of GQDs appears at about 460 nm, indicating that GQDs transfers energy to Eu$^{3+}$ after absorbing excitation light. Decay curves of the samples (i) and (iii) are shown in Fig. 3c, which can be fitted with linear functions [here, the nonlinear equation of \( y = A_1 \exp(-x/t) + y_0 \) is transformed into the linear equation of \( \ln(y - y_0) = A_1 - x/t \)]. Lifetimes of the samples (i–iii) obtained from the curves are 5.22, 5.90 and 6.28 ms, respectively. Fig. 3d shows chromaticity diagrams of the samples (i–iii), and corresponding CIE chromaticity coordinates are as follows: (i) (0.6156, 0.3839); (ii) (0.6150, 0.3845); (iii) (0.6147, 0.3848). Fig. 3d further shows that the three points have overlapped as one point, indicating the emissions of the samples have almost the same red color.

3.3 Luminescent properties at different concentrations of Eu$^{3+}$, Yb$^{3+}$ and GQDs

Fig. 4a shows PL spectra of KGF:xEu$^{3+}$. All KGF:xEu$^{3+}$ samples show similar PL spectra with different relative intensities. Three strong peaks observed at 591, 611, 650 and 700 nm are attributed to the $^5D_0 \rightarrow ^7F_1$, $^5D_0 \rightarrow ^7F_2$, $^5D_0 \rightarrow ^7F_3$ and $^5D_0 \rightarrow ^7F_4$ transitions of Eu$^{3+}$ ion, respectively. In addition, the weak peaks at 533, 552, 578, and 625 nm (shoulder peak) belong to the $^5D_1 \rightarrow ^7F_1$, $^5D_1 \rightarrow ^7F_2$, $^5D_1 \rightarrow ^7F_3$ and $^5D_1 \rightarrow ^7F_4$, respectively. Curve of the emission intensity at 592 nm changes with different x is a parabola, achieves its summit when x is about 0.20.

Fig. 4b shows the emission spectra (PL) of KGF:0.20Eu$^{3+}$, yYb$^{3+}$ samples. The dependent curve of emission intensity at

Table 2  Lattice parameters and volumes of samples (i–iii)$^a$

| Sample | \(a = b = c/\text{Å}\) | Vol./Å$^3$ |
|--------|----------------|----------|
| (i)    | 5.748          | 189.93   |
| (ii)   | 5.740          | 189.18   |
| (iii)  | 5.751          | 190.23   |

$^a$ (i) KGF:0.20Eu$^{3+}$, (ii) KGF:0.20Eu$^{3+}$, 0.002Yb$^{3+}$, (iii) KGF:0.20Eu$^{3+}$, 0.002Yb$^{3+}$@GQDs$_2$ mg mol$^{-1}$.

Fig. 2  SEM image, EDS, XPS and FTIR spectra of sample (iii), KGF:0.20Eu$^{3+}$, 0.002Yb$^{3+}$@GQDs$_2$ mg mol$^{-1}$: (a) SEM, (b) EDS, (c) XPS, (d) FTIR.
592 nm on different y is also a parabola, and it has a maximum when x is about 0.002.

Based on the result of Fig. 4b, the optimal double-doped sample (KGF:0.20Eu$^{3+}$, 0.002Yb$^{3+}$) was further coated with GQDs. The emission spectra (PL) of KGF:0.20Eu$^{3+}$,0.002Yb$^{3+}$@GQDs$_{z}$ samples are shown in Fig. 4c. Fig. 4c shows that the PL intensity curve with different z is a parabola with a maximum value. The optimal concentration of GQDs for KGF:0.20Eu$^{3+}$,0.002Yb$^{3+}$@GQDs$_{z}$ phosphors is 2 mg mol$^{-1}$. The results show that the obvious luminescent
enhancement can be obtained by coating a small amount of GQDs.

### 3.4 Luminescent thermal stabilities

Luminescent thermal properties of the samples (i–iii) at different temperatures are shown in Fig. 5. Fig. 5a–c shows the integrated PL intensity curves of the three samples are strongly affected by temperature. Fig. 5d shows that the curve of the sample (i) is a nonlinear curve with monotonically decreasing, and curves of the sample (ii–iii) are parabolas, which is caused by the NTQ effect. Fig. 5d further shows that curve (iii) is higher than curve (ii), indicates that the effect of the sample (iii) is stronger than that of the sample (ii). The highest points of the two parabola curves (curves (ii) and (iii)) are 130, 170 °C, which the corresponding integrated PL intensities are 117.7 and 156.5% of that initial value at 30 °C, respectively. So, the optimal sample (KGF:0.20Eu3+, 0.002Yb3+@GQDs2 mg mol⁻¹) has high luminescent thermal stability, due to the NTQ effect.

The results of Fig. 5d show that: (a) the sample (i) has no such the effect, but the sample (ii) has, indicating that the effect is induced by co-doping of Yb3+. The effect induced by double doped ion was also observed in LuVO4:Bi3+, Eu3+ phosphor reported by Zhang et al.; (b) the effect is further obviously enhanced when the surface of the double doped sample is coated with GQDs. The enhancement of the effect by GQDs was also occurred in GQDs coated Mn4+ doped fluoride red emitting phosphor.

In previous work, we have reported that KGdF4:Eu3+, Sm3+@GQDs also has the NTQ effect, but their co-doping sample (KGdF4:Eu3+,Sm3+) has not. The results show that Eu3+ and different rare earth ions co-doped on the same matrix (KGdF4) have different luminescent thermal properties, and only proper co-doping can obtain samples with the NTQ effect. Mechanism of the effect has been discussed in some reports, and was suggested that the electrons of phosphor gain energy supply from the electron trap produced by the defect of the matrix with the increase of temperature, and then the effect is induced.

In this paper, the mechanism of the effect is further explored from the perspective of energy conservation. The intensity of the excited light is constant whether at the parabola starting point or at its apex (Fig. 5d). However, the integrated PL intensity at the apex is stronger than that at the starting point. According to the mechanism reported in the literature, the extra energy required at the apex is provided by the electrons in the electron trap. From a macro point of view, in the whole variable temperature test system, the only way to provide extra energy is the thermal energy of the heating system. Therefore, it is not difficult to find that the energy in the electron trap is converted from the thermal energy. This mechanism can be represented with Fig. 5e and f. Fig. 5e and f show that: first, after double-doping of Yb3+ (Fig. 5e), the defects suitable for forming electron traps are formed in the matrix, and then the electrons are driven by heat and trapped by the defects, and finally the electrons transfer energy to the 5L6 energy level of Eu3+ to resulting in the NTQ effect. In addition, when the double-doped sample is coated with GQDs (Fig. 5f), more defects are produced and more electrons are captured, so more energy is transferred to the 5L6 energy level, which induces a stronger the NTQ effect. The mechanism of Fig. 5e and f is suggested as mechanism of energy conversion from heat to light.

### 3.5 Luminescent performances of prototype red light LEDs

The luminescent performances of the prototype red light LEDs are shown in Fig. 6. The LEDs were fabricated by coating mixture of the samples [i, iii] KGF:0.20Eu3+, 0.002Yb3+@GQDs2 mg mol⁻¹.
and epoxy onto a 395 nm UV chip and tested at different driving currents. Fig. 6a shows the electroluminescence spectra of the LEDs, which consist emissions of UV chip and the samples. Fig. 6b depicts CIE chromaticity diagrams of the LEDs, which are (0.5373, 0.3494) and (0.5163, 0.3392), respectively. Results of Fig. 6b indicate that the LEDs emit orange red light, which is supported by orange main emission (about 592 nm) in Fig. 3b.

The red color ratio of the LEDs at different driving currents is shown in Fig. 6c. Fig. 6c illustrates that the trend of the two curves is just the opposite. Curve (iii) increases with the increase of the driving current, while curve (i) decreases with the increase of the driving current. It is well known that the operating temperature of the chip increases with the increase of the driving current of the chip. It can be seen from the illustration in Fig. 5d, before 170 °C, curve (iii) monotonically rises with the increase of temperature, while curve (i) monotonically decreases with the increase of temperature. Obviously, the results of Fig. 6c are supported by results of Fig. 5d, and it can be further inferred that the operating temperature of the UV chip driven by 50 mA current does not exceed 170 °C. Furthermore, the results of Fig. 6c also indicate that the optimal sample (KGF:0.20Eu3+, 0.002Yb3+@GQDs2 mg mol¹) has high luminescent thermal stability, which can be applied for high power WLEDs.

Fig. 6 Luminescent performances of prototype LEDs (samples (i, iii) + 395 nm UV chip), (i) KGF:0.20Eu3+, (iii) KGF:0.20Eu3+, 0.002Yb3+@GQDs2 mg mol¹: (a) electroluminescence spectrum under a 20 mA drive current, (b) CIE chromaticity diagram, (c) red ratio under a different drive current.

4. Conclusions

In summary, a series of KGdF4:Eu3+,Yb3+@GQDs red emitting phosphor have been synthesized by co-precipitation method, and their luminescent thermal properties have also been studied. More intriguingly, the NTQ effect is formed induced by the double doping of Yb3+, and the effect is further enhanced by GQDS coating. The strongest integrated PL intensities of the optimal double doped sample and the optimal GQDs coated sample are at 130 and 170 °C, which the corresponding integrated PL intensities are 117.7 and 156.5% of that initial value at 30 °C, respectively. The NTQ effect makes the optimal GQDs coated sample have good luminescent thermal stability. A mechanism of energy conversion from heat to light is suggested for the NTQ effect.

Conflicts of interest

The authors declare that they have no conflict of interest.

Author contributions

Zhigao Wu: methodology, formal analysis, investigation, writing – original draft. Chang Chen: investigation. Yaxiong
Wang: investigation. Chaolian Luo: investigation. Sen Liao: conceptualization, supervision. Yingheng Huang: review & editing, visualization. Junyu Ming: review & editing, visualization.

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