Asymmetric MoS$_2$/Graphene/Metal Sandwiches: Preparation, Characterization, and Application

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Atomically thin semiconductors, monolayers of transition-metal dichalcogenides (TMDCs), are of increasing interest as possible substitutes for Pt in the catalysis of hydrogen evolution reactions (HERs). One of the most studied members of the TMDC family is molybdenum disulfide (MoS$_2$), used as a catalyst in HER because of its low cost, high chemical stability, and good electrocatalytic properties. Unfortunately, bulk MoS$_2$ has poor conductivity, which is attributed to the lateral transfer of electrons along the layered structure of MoS$_2$ nanosheets. Graphene (GR), the first isolated single atomic layer, composed of sp$^2$-hybridized carbon atoms, has good conductivity, high chemical durability, and a large surface area. For this reason, it has become an important substrate for composite materials in catalysis research and energy storage devices.

The different species of 2D layered materials can be stacked to form heterostructures, producing novel physical and chemical properties. There have been significant prior studies demonstrating fabrication of MoS$_2$ and GR composites to enhance the catalysis of the most versatile energy storage system and energy carrier, i.e., hydrogen gas. These 2D layered materials are excellent candidates for applications in electrochemical energy storage because of their high available surface area and versatile electronic structure.

The question of the relative chemical activity of basal planes compared to the much more active edge planes (primarily in the case of GR) is still an open subject, as quantifying differences in edge dimensions is challenging due to the inherently high surface energy. Liquid/liquid self-assembly techniques consist of the dispersion of nanoparticles (NPs) in a hydrophilic or hydrophobic phase, followed by the driving of NPs to the oil/water interface using mechanical agitation or with the assistance of an inducing agent. The interface between two immiscible electrolyte solutions (ITIES) is a special form of the generic liquid/liquid interface, where the presence of electrolyte in both phases provides external control over charge transfer, such as ion, electron, and proton-coupled electron transfers. GR self-assembly at various organic/water interfaces has received much attention recently, as the assembled films offer the possibility of straightforward transfer, with the specific aim of preparing transparent or modified electrode materials. Self-assembly of WSe$_2$ flakes has also been demonstrated at the interface of two immiscible nonsolvents (ethylene glycol and hexane). However, to date, there is a lack of ordered assembly of layered TMDCs or their composites at the ITIES. In fact, only a small body of work has considered the construction of novel composites at the ITIES, i.e., metal-NP-decorated, low-dimensional carbon materials and bimetal GR sandwiches. Moreover, functionalization of the single-layer GR and preparation of the sandwiched structures, such as building up modified layers by adding them on top of one another to form doped GR papers or layered structures, altering topology as well as electronic structure, have also been reported recently.

Here, we have fabricated a range of asymmetrically dual-functionalized GR-based nanoclusters, including layered MoS$_2$ and metal NPs on the top-side and metal NPs on the under-side. The resulting semiconductor/GR/metal sandwich structures have been characterized using a wide range of microscopic and spectroscopic techniques, including 3D electron tomography. We report three- and four-electrode configuration electrochemical measurements using the resultant nanocomposites to demonstrate the capability of the heterostructures for HER and enhanced capacitance performance.

The various MoS$_2$/GR heterostructures were obtained from either chemical vapor deposition (CVD)-grown, or liquid-phase exfoliated MoS$_2$ flakes and CVD-grown GR monolayers. The vertically aligned CVD-grown MoS$_2$ nanosheets (hereafter, vMoS$_2$) and the liquid-phase exfoliated MoS$_2$ flakes (hereafter, eMoS$_2$) were used to decorate the top-side of the GR, forming the vMoS$_2$/GR and eMoS$_2$/GR nanocomposites, respectively (Figure 1). The eMoS$_2$ flakes were assembled at the ITIES using ultrasonic agitation, while the vMoS$_2$ nanosheets were transferred from the Si/SiO$_2$ (the substrate for growth) to aqueous solution through etching of the Si/SiO$_2$ substrate by potassium hydroxide (KOH) solution. The eMoS$_2$ and vMoS$_2$ nanosheets were “fished out” from the organic/water or water/air interfaces, respectively, using CVD GR supported on a copper foil.
More details about the preparation and transfer of the MoS$_2$/GR structures are provided in the Experimental Section and in the Supporting Information (Section S1 and S2).

After transferring the vMoS$_2$ nanosheets to the single-layer GR, their well-aligned layered structure is maintained. Each of the layers consist of three covalently bonded atomic sheets, i.e., S–Mo–S (Figure 1a), with a layer-to-layer distance of about 6 Å, in agreement with literature values.\[10\] X-ray photoelectron spectroscopy (XPS) spectra of the vMoS$_2$ nanosheets are presented in Figure 1c,d, the position and shape of the Mo 3d and S 2p doublets confirm that the MoS$_2$ is present exclusively as the semiconducting 2H phase.\[18\] The apparent shoulder present on the Mo$^{4+}$ 3d$^{3/2}$ peak (Mo$^{6+}$ 3d$^{5/2}$, green peak at \(\approx 233\) eV in Figure 1c), and another component, Mo$^{6+}$ 3d$^{3/2}$ at \(\approx 235\) eV, are attributed to mild oxidation of the molybdenum ions (Mo$^{4+}$ to Mo$^{6+}$)\[2a,b\] (additional XPS surface analysis can be found in Section S7 in the Supporting Information). The electron microscope images of the eMoS$_2$/GR structures in Figure 1b indicate a less uniform coverage of the eMoS$_2$ on the monolayer GR compared to the vMoS$_2$/GR nanocomposite. The variable thickness of the eMoS$_2$ flakes can be seen, with flake thicknesses estimated to be mostly 3–10 layers, as illustrated by Figure 1b. The Raman spectra of the eMoS$_2$/GR (black curve) and vMoS$_2$/GR (red curve) heterostructures are shown in Figure 1e. The expected peaks’ characteristic of single-layer GR are observed: the D peak that corresponds to edges and defects, the G peak that is associated with the $E_{2g}$ vibration mode of the $sp^2$ bonds, and the 2D peak that is caused by the scattering of two phonons which can occur independently of defects. The inset in Figure 1e highlights the presence of peaks corresponding to the most intense phonon modes of MoS$_2$, including the in-plane ($E'_{2g}$) and out-of-plane ($A_{1g}$) lattice vibrations,\[19,20\] which exhibit hardening and softening, respectively. The Raman spectra from the GR on the under-side of both kinds of MoS$_2$ material showed upshifts of both the G and 2D bands. In the case of the G peak, the increase is observed from 1585.5 cm$^{-1}$ (pristine CVD GR) to 1587.2 cm$^{-1}$ (eMoS$_2$/GR), and 1587.5 cm$^{-1}$ (vMoS$_2$/GR), while the 2D band is upshifted from 2677.1 cm$^{-1}$ (pristine CVD GR) to 2687.8 cm$^{-1}$ (eMoS$_2$/GR), and 2697.3 cm$^{-1}$ (vMoS$_2$/GR). The width of the 2D band increases from 30.4 cm$^{-1}$ (pristine CVD GR) to 40.7 cm$^{-1}$, and 43.5 cm$^{-1}$ for eMoS$_2$/GR and vMoS$_2$/GR heterostructures, respectively. The frequency upshifts of the G and 2D bands combined with the increase in 2D width indicate p-type doping,\[21\] which can be explained by interlayer coupling between GR and MoS$_2$,\[22\] and photoinduced electron transfer...
from MoS$_2$ to GR\cite{23} due to excitation by the visible light (532 nm) irradiation of the Raman laser. When comparing the Raman spectra of eMoS$_2$/GR (black) and vMoS$_2$/GR (red), the difference in the intensity ratio ($I_{2g}/I_{1g}$) is clearly evident, the ratio for the eMoS$_2$/GR sample is $\approx 0.49$ while the ratio for the vMoS$_2$/GR sample is $\approx 0.24$, verifying that the vMoS$_2$/GR sample is indeed edge-terminated and vertically aligned\cite{10,24} (detailed comparison of the vMoS$_2$ and basal planes is given in Section S3 in the Supporting Information).

Considering the band structure difference between the indirect-bandgap semiconductor of bulk MoS$_2$ to the direct-bandgap semiconductor in monolayer MoS$_2$, we would not expect to see any photoluminescence (PL) in the case of the vMoS$_2$/GR heterostructure, which was true for the samples supported on the Si/SiO$_2$ wafer before transfer to the GR. Surprisingly, PL appeared when measuring free-standing vMoS$_2$/GR floating on the surface of deionized water. The optical micrograph (Figure 1f) displays this floating vMoS$_2$/GR, with “singly” coated ($\approx 50$ nm), folded nanosheets ($\approx 150$ nm), and the exposed part (top of image) showing the GR support (characterization of these height profiles is given in Section S3 in the Supporting Information). The PL was strongest for the exposed GR surface (black curve in Figure 1g), indicating breakage of some thin (mono- or bilayer) vMoS$_2$ nanosheets during its transfer and subsequent heterostructure formation. The negligible quenching of the observed strong PL response indicates that the interlayer charge transfer, which results from the interaction between the MoS$_2$ and GR sheets, is relatively weak and is similar to what has been observed for mechanically stacked heterostructures\cite{23}. This is in contrast with similar CVD-grown heterostructures where strong quenching of the PL from the MoS$_2$ grown on GR was observed\cite{23}.

The pristine and MoS$_2$-decorated single-layer GR can be functionalized using metal NPs on either both sides, or solely on the under-side. A number of MoS$_2$/GR/metal sandwiches have been prepared using a polarizable organic/water interface.

Three kinds of functionalization of the GR layers were carried out. First, top-side decoration was performed using semiconductors (eMoS$_2$, vMoS$_2$) or metal NPs (Pd, Pt, Au, Ag). Second, under-side decoration of metal NPs achieved at the organic/water interface using either the spontaneous deposition route (Pd, Au, Ag), or electrochemical deposition (Pd, Pt), or by the controlled etching of the Cu foil used for the CVD growth of GR (Cu). In the latter special case, the exact amount of Cu NPs remaining on the GR could be precisely determined by control of the etching conditions (Supporting Information, Section S7). Third, bi-facial decoration (asymmetric or symmetric) was achieved by decorating the GR on the top-side, before the decorated GR-based hybrid is subsequently functionalized on the under-side. Hereafter, the resultant nanocomposites are termed to signify the relative positions of the species to the GR layer, e.g., if the top- and under-sides of the GR are decorated with vMoS$_2$ nanosheets and Pd NPs, respectively, the name of the nanocluster will be vMoS$_2$/GR/Pd. Examples of several dual-decorated CVD GR-based nanostructures, such as vMoS$_2$/GR/Pd (top-part of figure), eMoS$_2$/GR/Pd (from the left to right in the bottom of figure), Pt/GR/Cu, and Pd/GR/Pd, are shown in Figure 2.

The functionalization of CVD GR with metal NPs at an interface has been described previously\cite{16,25}, but a description is also

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**Figure 2.** 3D illustration of the vMoS$_2$/GR/Pd sandwich (top). Schematics of the eMoS$_2$/GR/Pd, Pt/GR/Cu, and Pd/GR/Pd sandwiches (bottom of image, from left to right).
given here in the Experimental Section. Briefly, the MoS$_2$/GR heterostructures are assembled at the polarizable organic/water interface, then the hybrid material is transferred to the surface of an aqueous layer/phase before a lower density organic phase is layered on top. The resulting ITIES offers the possibility to induce metal-NP deposition on the assembled GR using metal precursors in the aqueous phase and an organic phase electron donor, employing either simple galvanic displacement reactions (as a function of potential difference) or direct potential control. The MoS$_2$/GR at the interface was functionalized with Pd NPs by adding tetrachloropalladate (PdCl$_4^{2−}$) in the aqueous phase and decamethylferrocene (DecMFc) as the electron donor agent in the organic phase respectively (the cell composition is shown in Composition 1).

High-angle annular dark-field (HAADF) scanning transmission electron microscopy (STEM) images show the vMoS$_2$/GR/Pd sandwich, visualizing the vertically aligned MoS$_2$ structure and the bright Pd NPs (Figure 3b). The size distribution of individual Pd NPs is ~5–10 nm (Figure 3c) and these individual NPs assemble into “snowflake-like” structures (~100–200 nm) (Figure 3a), respectively. Confirmation that the “snowflake-like” structures correspond to Pd (green) and the stacked flakes correspond to Mo (red) is provided by STEM energy-dispersive X-ray (EDX) spectrum imaging (Figure 3d). Further confirmation of the presence of both Pd and Mo$_2$S$_4$ is provided through electron diffraction patterns (DPs) (inset of Figure 3b), displaying polycrystalline rings characteristic of Pd and Mo$_2$S$_4$. Moreover, a HAADF-STEM tomography reconstruction of a small area (Figure 3e) shows the vertical displacement of the Mo$_2$S$_4$ and Pd layers, where the different materials can be segmented due to their distinct morphologies. This reconstructed area is densely decorated by Pd NPs, showing clearly the “sandwich structure” of the vMoS$_2$ nanosheets on the top-side (Figure 3f) and the Pd “layer” on the under-side (g); the scale bars correspond to 100 nm. The interfacial deposition time was 1 min of Pd NPs decoration to under-side of GR.

The same reaction was undertaken on the under-side of the eMoS$_2$/GR structure using PdCl$_4^{2−}$, AuCl$_4^{−}$, and Ag$^{+}$ in the aqueous phase, resulting in eMoS$_2$/GR/Pd, eMoS$_2$/GR/Au, and eMoS$_2$/GR/Ag sandwiches, respectively (further characterization is given in Section S5 in the Supporting Information). The other route to decorate the interface-assembled heterostructures is via potentiodynamic metal-NP deposition (using cyclic
This cell contained either PdCl$_4^{2-}$ or PtCl$_4^{2-}$ in the aqueous phase and 1,1'-dimethylferrocene (DMFc) in the organic phase (Composition 2).

In this case, DMFc acts as the electron donor, and during its electro-oxidation through the assembled MoS$_2$/GR heterostructures at the ITIES, the reduction of the PdCl$_4^{2-}$ to Pd NPs occurs at the aqueous side of the interface on the under-side of the MoS$_2$/GR, by analogy with the similar Pd electrodeposition on the pristine CVD GR assembled at the ITIES.$^{[28]}$ The potential cycling for the electrodeposition is given in Section S6 in the Supporting Information. More evidence for the asymmetrically dual-decorated GR sandwiches can be found in the videos of the aligned HAADF-STEM tilt series’ and final reconstructions provided as Supporting Information (Movie S1 and S2).

Electrochemical measurements using microdroplets$^{[9a,c,29]}$ were performed in order to investigate the HER catalytic performance and to determine the interfacial capacitance for the different GR-based composites. In order to assess the catalytic activity toward HER, polarization curves of five samples (GR, vMoS$_2$, vMoS$_2$/GR, eMoS$_2$/GR/Pd, and vMoS$_2$/GR/Pd) were measured in aqueous droplets containing 0.1 M HCl and 6 M LiCl (Figure 4a).

The Tafel slopes estimated from the linear portions of the polarization curves (Figure 4a) using the Tafel equation (Equation S2, Section S10, Supporting Information) for all the studied samples (Figure 4b). The edges sites are believed to be the most catalytically active and thus one would expect the vMoS$_2$ to naturally have superior performance.$^{[1b]}$ Indeed as predicted, vMoS$_2$ exhibits 112.0 ± 0.99 mV dec$^{-1}$ at a low overpotential, the current density is high, with the smaller Tafel slope corresponding to a faster increase of HER rate.$^{[10]}$ While, the vMoS$_2$/GR/Pd nanocomposite exhibits superior activity with the smallest Tafel slope value of 71.0 ± 0.93 mV dec$^{-1}$ at the lowest overpotential. However, lower Tafel slopes were reported previously,$^{[2a,10]}$ e.g., those of the electrochemical tuned vertically aligned MoS$_2$ nanofilms (44 mV dec$^{-1}$)$^{[2a]}$ or MoS$_2$/reduced GR oxide modified glassy carbon electrode (41 mV dec$^{-1}$).$^{[7a]}$ In acidic media there are three operative reaction mechanisms$^{[31]}$ for the HER: first the initial discharge step (Volmer reaction where the Tafel slope, $b$ is around 120 mV dec$^{-1}$), which can be followed by either an electrochemical desorption step (Heyrovsky reaction, $b$ ≈ 40 mV dec$^{-1}$) or a recombination step (Tafel reaction, $b$ = 30 mV dec$^{-1}$). Considering the 71.0 mV dec$^{-1}$ Tafel slope found for the vMoS$_2$/GR/Pd nanocomposite, it is likely the combination of initial discharge and electrochemical desorption steps could be the rate-limiting processes. A recent report has discussed the use of the chemically exfoliated MoS$_2$ nanosheets, and the relative efficiency of hydrogen evolution from the 1T and 2H phases of the MoS$_2$ nanosheets.$^{[32]}$ The reported superior electrocatalytic activity is obtained only in the case of the high concentration of 1T MoS$_2$ particles, while the semiconducting 2H phase’s activity for HER reaction can be improved by increasing its conductivity through doping with SWNTs, we note that Tafel slopes similar to those reported here were found for the 2H case.

The capacitive performance of the GR hybrid films was electrochemically tested to determine the electric double-layer capacitances ($C$) for each sample (calculated using Equation S3, Section S10, Supporting Information). The specific capacitance ($C_{sp}$) values were calculated using the $C$ and surface area.
values. The dependence of the $C_{sp}$ on scan rate is depicted in Figure 4c, showing the well-known inverse correlation with increasing scan rate due to the transport limit on the supply of ions. At 100 mV s$^{-1}$ the calculated specific capacitance for the vMoS$_2$/GR/Pd (10.7 mF cm$^{-2}$) is $\approx$ 9, 130, 270, and 2400 times larger than that for vMoS$_2$/GR (1.21 mF cm$^{-2}$), eMoS$_2$/GR/Pd (81.2 µF cm$^{-2}$), vMoS$_2$ (38.9 µF cm$^{-2}$), and GR (4.4 µF cm$^{-2}$), respectively (Figure 4c). An ideal electric double-layer capacitor supercapacitors should possess extremely high stability with minimal degradation in performance with repeated charge/discharge cycles, while pseudocapacitors typically sacrifice cycling stability for increased energy density. Figure 4d shows the cycling stability of the vMoS$_2$/GR/Pd nanocluster over 300 cycles. There is a large increase ($\approx 170\%$) when comparing the $C_{sp}$ values precycling relative to postcycling. A prolonged cycling in order to check both electrical and mechanical stabilities up to 2000 cycles has shown durability of the vMoS$_2$/GR/Pd nanocomposite and further increase in capacitance ($\approx 340\%$) observed. The heterostructures appear to be quite robust during the timescale used for measurements, although further long term studies are required to ascertain their stability before practical applications are considered. This increased capacitance with continuous cycling has been reported previously for GR and MoS$_2$ composites, and was also attributed to the partial exfoliation with continued ion intercalation/deintercalation increasing the active surface area, termed “electroactivation.” In the particular case of a dual-decorated GR sandwich (vMoS$_2$/GR/Pd), during the “electroactivation,” the Li$^+$ ions can also penetrate through and incorporate into the porous metal nanostructures.

Both carbon nanotubes and GR assemblies at the ITIES, as well as exfoliated TMDCs in the aqueous phase, have received significant interest for applications in energy storage and conversion, mainly HER and the oxygen reduction reaction. Considering the high HER activity measured and the enhancement in the case of the vMoS$_2$/GR/Pd, significant proton transfer through the GR single-layer must occur, either through defects or via the intrinsic proton conductivity, in order to explain the enhanced catalytic and capacitive performances of the under-side functionalized composites (with Pd NPs on the under surface) versus the top-side only decorated composites. We have studied the catalytic role of the as-prepared dual-decorated GR-based composites in HER at polarized 6 M LiCl (aq)/DCB interfaces (Composition 3), between an aqueous acidic solution and a solution of DecMFc acting as a molecular electron donor in the organic phase (Figure 5). The reaction between DecMFc and aqueous H$^+$ catalyzed by adsorbed GR-based hybrid particles is depicted in the schematic (Figure 5a).

In the case of HER at the ITIES, the catalytic process occurs at high Galvani potential differences (which is defined as the potential difference between the aqueous and organic phase by convention), and not at negative reducing potentials as on a conventional solid electrode. The observed redox peaks between $-0.1$ V and $-0.3$ V correspond to the DecMFc$^+$ cation transfer, while the more positive redox activity (further than 0.0 V) corresponds to the HER (more details are given in Section S9).

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**Figure 5.** a) Schematic of the HER at the interface. b–e) Cyclic voltammograms at 6 M LiCl(aq)/DCB interface, comparing Pd/GR (b), Pt/GR (c), eMoS$_2$/GR (d), and vMoS$_2$/GR (e) heterostructures assembled at the interface, using a scan rate of 50 mV s$^{-1}$. 

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in the Supporting Information). In order to investigate the as-prepared dual-decorated composites to enhance the HER at the 6 M LiCl/DCE interface, the Pt/GR, Pd/GR, eMoS₂/GR, and the vMoS₂/GR-based heterostructures (under-side was functionalized with residual Cu particles) were compared in Figure 5b–e, respectively. Furthermore, the assembly of individual, macroscopic (cm² scale) GR-based catalyst material at the ITIES containing redox active species for HER minimizes the possibility of “uncontrolled adsorption” or sedimentation of the exfoliated TMDCs particles at the interface.

For all cases, the top-side decorated sample, e.g., vMoS₂/GR (red curves) was compared to the dual-functionalized, e.g., vMoS₂/GR/Pd (blue curves) and the GR/Cu composites (green curves), employing the GR/Cu as a standard to compare with the top-side decorated layered semiconductors (eMoS₂ or vMoS₂) composites and the both sides decorated ones, respectively. In the case of the under-side functionalized nanocomposites (e.g., GR/Cu), the protons are adsorbed on the catalyst surface on the under-side of GR, which conducts the electrons from the DecMFC oxidation to the protons, leading to fast electron transfer (Δ φ w = +0.183 V). Employing the top-side decorated hybrids, the protons have to transfer to the catalyst surface on the top-side through the GR single-layer [35] and reduction must occur there (e.g., Δ φ w = −0.109 V for Pt/GR). However, in the case of the asymmetrically dual-functionalized composites, the protons can be either adsorbed on the under-side of the catalyst and reduced there, or transfer to the top-side of the catalyst through the monolayer GR, or do both leading to an enhanced activity: Δ φ w ≈ −0.022 V and Δ φ w ≈ −0.091 V for eMoS₂/GR/Cu and vMoS₂/GR/Pd (close to the Pt/GR/Cu composite, where Δ φ w = −0.123 V), respectively. Previous literature has reported a range of catalysts employed to increase the rate of HER reaction at the ITIES, including Pt, Pd,[16] and Cu[17] NPs. Proton reduction can occur either in the aqueous or in DCE phase, the proton reduction potential in the DCE phase has been reported as −0.55 V.[18] Herein, the enhanced activity is indicated with the lower overpotential values for HER at the interface. It is the combination of the highly catalytically active palladium combined with the MoS₂ which leads to the excellent electrocatalytic performance of the composite for HER. Considering the deeper understanding of the effect of GR-based nanocomposites for the different catalysis, further investigations and the isolation of the operative factors will be the subject of additional studies, such as the investigation of the different metal NPs with different behaviors and comparison of several conditions (pH, different thickness of the heterostructures, and varying their compositions).

In summary, we have synthesized and characterized various MoS₂/GR heterostructures, including CVD-grown and liquid-phase exfoliated MoS₂ nanosheets and flakes, respectively. In our work, the interlayer coupling between the MoS₂ and GR would be expected to be most similar to the mechanically stacked structures which also show very little quenching.[23] These hybrid heterostructures were functionalized with metal NPs (Au, Ag, Pd, Pt, Cu) on the opposite side to the MoS₂, forming asymmetric dual-side decorated GR sandwiches, via both spontaneous and potential-controlled procedures.

The different nanocomposites show diverse catalytic activities, with the best Tafel slope for the HER found to be 71.0 mV dec⁻¹ for the vMoS₂/GR/Pd sandwich, outperforming the other MoS₂ containing composites (161.2–196.5 mV dec⁻¹). This vMoS₂/GR/Pd sandwich has exhibited enhanced catalytic activity compared to both the pristine materials and the other heterostructures studied in this work. Considering the ~200% and ~340% increase during the cycling stability measurements in the case of 300 and 2000 cycles, respectively, this hybrid structure is also a promising candidate as a supercapacitor electrode. Additionally, the dual-decorated GR sandwiches, mainly the vMoS₂/GR/Pd hybrid, demonstrated enhanced HER activity at a 6 M LiCl/DCE interface.

The key finding is the creation of a unique platform of asymmetrically functionalized MoS₂/GR heterostructures, and the investigation of fundamental electrocatalytic properties of these structures. Furthermore, this work leads to new prospects for assembly of liquid-phase exfoliated layered 2D semiconductors at polarizable organic/water interfaces, offering the platform for developing fundamental studies and exploiting them in energy storage and conversion, e.g., lithium intercalation and photoelectrochemical hydrogen production. Moreover, this opens up further, new vistas for GR asymmetric functionalization including composite structures incorporating 2D semiconductors, metal NPs, or materials with switchable redox systems, i.e., conducting polymers. Through inclusion of two different catalyst materials on each side of the GR single-layer, the asymmetrically dual-decorated GR-based composites provide an ideal platform to catalyze several biphasic reactions, for example an enhancement of two different redox reactions simultaneously occurring in the organic and aqueous phases respectively.

**Experimental Section**

**Preparation of eMoS₂/GR and vMoS₂/GR Heterostructures:** The MoS₂ dispersions (eMoS₂) were created by liquid phase exfoliation and the large-area, thickness-controlled, vertically aligned MoS₂ nanosheets (vMoS₂) were prepared by sulfurization of thin metal films (CVD method), as reported previously.[39] The detailed preparation procedures for both kinds of MoS₂ material, together with the UV-vis characterization and concentration calibration for eMoS₂ solutions are provided in the Supporting Information (Section S2 and S3).

The eMoS₂ flakes’ assembly at the organic/water interface was established according to the procedure for few-layer GR flakes’ assembly, as described elsewhere.[19] Briefly, an aliquot of the eMoS₂ dispersion (20 µg mL⁻¹) with the organic electrolyte solution in DCE was placed in contact with an equal volume of aqueous phase, and then self-assembly of the eMoS₂ flakes at the interface was achieved using ultrasonic agitation (15 min). Both phases contained supporting electrolytes, 0.1 M LiClO₄ in the aqueous phase and 0.1 M TBAClO₄ in the organic phase, respectively. After 2 h, the eMoS₂ flakes assembled at the interface in a thin film. Subsequently, a part of this film was “fished out” using the CVD GR lying on the Cu foil and the sample was dried, forming the eMoS₂/GR structures (Figure S2, Supporting Information).

The vMoS₂ thin sheets were transferred to CVD GR to form the vMoS₂/GR heterostructure (Figure S7, Supporting Information). The transfer was carried out by spin-coating the vMoS₂ sheet with 3% 950 K poly(methyl methacrylate) (PMMA) solution (in anisole) on the Si/SiO₂ substrate at 3000 rpm for 60 s using SPIN2000-NPP spin processor (SPS-Europe B.V.). This resulted in the formation of a PMMA/vMoS₂/Si/SiO₂ substrate multilayer structure (Figure S7b, Supporting Information). The Si/SiO₂ was etched from underneath the PMMA/vMoS₂/multilayer by 2 µ KOH (Figure S7c, Supporting Information), then the SiO₂ between the MoS₂ and the Si substrates was completely etched, leaving...
PMMA/vMoS$_2$ multilayers floating on the KOH surface (Figure S7d, Supporting Information), and then cleaned three times by transfer to pure water, resulting in a pure PMMA/vMoS$_2$ (Figure S7e, Supporting Information). The remaining PMMA/vMoS$_2$ heterostructure was then “fished out” with an Si/SiO$_2$ wafer, and then cleaned three times by transfer to pure water, resulting in a PMMA/vMoS$_2$/GR stack (Figure S7f, Supporting Information), after drying the sample the Cu foil was etched from underneath the PMMA/vMoS$_2$/Cu foil stack (Figure S7g, Supporting Information) using an Si/SiO$_2$ wafer.

**Preparation of Mono- or Bimetal Graphene Sandwiches:** The top-side decoration of the GR with Pd or Pt NPs was performed using a simple galvanic displacement for the nucleation of Pd or Pt NPs.$^{[14]}$ The underlying copper foil acted as an electron donor under the GR monolayer and transferred electrons to the noble metal. Alternatively, spontaneous Pd NP deposition on the under-side of the GR was achieved at the interface-assembled CVD GR. An interfacial redox reaction between decamethylferrocene (DecMFc), which is the organic phase electron donor, and PdCl$_2^-$ contained in the aqueous phase created GR/Pd layers.$^{[24]}$

**Characterization Techniques:** Characterization of samples was performed following either the preparation or functionalization of the MoS$_2$/GR sandwich. These composite layers were fished out using either an Si/SiO$_2$ substrate or specifically prepared free-standing samples supported on holey carbon-coated nickel Quantifoil grids (TAAB Laboratories Equipment Ltd.), dried in air for 15 min, then washed in an ethanol and isopropyl alcohol mixture (5 min) and then acetone (5 min), and finally blow-dried with nitrogen. To characterize the various GR-based samples, Raman spectroscopy (Renishaw inVia, 532 nm laser excitation), scanning electron microscopy (SEM) (Philips XL30 ESEM-FEG), atomic force microscopy (Bruker Multimode 8) measurements, and elemental analysis using XPS (Thermo Fisher Scientific, Inc.) and energy-dispersive X-ray spectroscopy (EDX in the SEM) were carried out. All of the measurements were obtained using the techniques at 3–5 different surface sites. Furthermore, HAADF-STEM imaging was performed on an FEI Talos F200X S/TEM with a high brightness X-FEG electron source and Super-X energy-dispersive silicon drift detectors. High-resolution EDX spectrum images were acquired in the Bruker Esprit software using the Talos’ Super-X detector system. A tilt series for HAADF-STEM tomography was acquired using Xplo3D acquisition software at angular increments of $1°$ from $−60°$ to $+60°$. Selected area electron diffraction patterns were acquired using a selected area aperture of $\approx 200$ nm diameter and a camera length of 840 mm. The displayed errors are either standard deviations (arithmetic averages of multiple measured values) or absolute errors determined from the best-fit errors.

**Applied Electrochemical Methods:** The electrochemical experiments were performed using either a three- or four-electrode configuration (Supporting Information, Section S4). In both cases the electrochemical signal was measured using a PCSTAT302N potentiostat (Metrohm Autolab). All the potentials reported in this work were plotted on the reversible hydrogen electrode scale,$^{[18]}$ calculated from Equation S1 (see Section S10, Supporting Information). Cyclic voltammetry was employed for the interfacial electrodeposition of Pd or Pt NPs on MoS$_2$/GR, redox reactions at the interface, and capacitance measurements. Additionally, the HER was investigated by linear sweep voltammetry.

**The Cell Configurations for the Experiments at the ITIES:** For voltammetry and $1 \times 10^{-3}$ M PdCl$_2^-$ (aq) $\mid 0.1$ M TBAClO$_4$ and $2 \times 10^{-3}$ M DecMFc$_{24}$ (Comp. 1)$^1$

**$\text{Ag/AgCl}_{(s)} | 0.1$ M LiCl and $1 \times 10^{-3}$ M PdCl$_2^-$ (aq) $\mid 0.1$ M TBAClO$_4$ and $2 \times 10^{-3}$ M DecMFc$_{24}$ (Comp. 1)**

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