Cementomimetics—constructing a cementum-like biomineralized microlayer via amelogenin-derived peptides

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Cementum is the outer-, mineralized-tissue covering the tooth root and an essential part of the system of periodontal tissue that anchors the tooth to the bone. Periodontal disease results from the destructive behavior of the host elicited by an infectious biofilm adhering to the tooth root and left untreated, may lead to tooth loss. We describe a novel protocol for identifying peptide sequences from native proteins with the potential to repair damaged dental tissues by controlling hydroxyapatite biomineralization. Using amelogenin as a case study and a bioinformatics scoring matrix, we identified regions within amelogenin that are shared with a set of hydroxyapatite-binding peptides (HABPs) previously selected by phage display. One 22-amino acid long peptide regions referred to as amelogenin-derived peptide 5 (ADP5) was shown to facilitate cell-free formation of a cementum-like hydroxyapatite mineral layer on demineralized human root dentin that, in turn, supported attachment of periodontal ligament cells in vitro. Our findings have several implications in peptide-assisted mineral formation that mimic biomineralization. By further elaborating the mechanism for protein control over the biominerallaboring, we afford new insights into the evolution of protein–mineral interactions. By exploiting small peptide domains of native proteins, our understanding of structure–function relationships of biomineralizing proteins can be extended and these peptides can be utilized to engineer mineral formation. Finally, the cementomimetic layer formed by ADP5 has the potential clinical application to repair diseased root surfaces so as to promote the regeneration of periodontal tissues and thereby reduce the morbidity associated with tooth loss.

Keywords: amelogenin; amelogenin-derived peptides; bioinformatics; biomineralization; cementomimetics; cementum; demineralization; remineralization

INTRODUCTION

The distinguishing characteristics of dental bioceramic hard tissues, e.g., enamel, cementum and dentin, are achieved due to the amount of the hydroxyapatite (HAp) crystallites formed, their nano- and microstructural characteristics and their overall architecture. These characteristics originate from the tissue-specific extracellular matrix (ECM) proteins found within each tissue during its development1–5 and which act as regulators of nucleation and growth of biological apatite.1–2,6 These ECM proteins, therefore, offer great potential in repairing mineralized tissues of the teeth damaged due to carries, trauma or periodontal disease. Periodontal disease is caused by the host’s inflammatory reaction to the bacterial biofilms that adhere to the tooth surface7 and results in periodontal tissue destruction. It is one of the most prevalent infections of mankind, and left unchecked, will result in the loss of the teeth and surrounding tissue, including bone.7–8 The use of dental ECM proteins as therapeutic agents to repair damage caused by such diseases has been limited so far due to technical and financial limitations in the identification, extraction and purification of these proteins. As the therapeutic use of native or recombinant ECM proteins for re-mineralization is not yet viable, research has focused on developing peptide mimics or identifying functional domains within these naturally occurring proteins.8–16 As a case study, we used the enamel protein, amelogenin, to identify similarity regions, i.e., domains with similar amino acid sequences, with a set of HAp-binding peptides (HABPs) that we previously experimentally selected.17 Amelogenin is expressed during tooth enamel formation and contributes to the hardest and most highly mineralized tissue in the human body by exerting control over the dimension and

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directionality of HAp crystals formed within the assembled protein matrix. 18–20 Although certain regions of amelogenin amino-acid sequence are proposed to interact with calcium phosphate minerals, an extensive mapping of putative crystal interaction or mineral nucleation domains is still lacking. 21–22 Here, we show that functional peptides can be derived from amelogenin using phage display-selected peptides as a knowledge base and identifying next generations of peptides by bioinformatics. We then show that these peptides can function to direct the cell-free re-mineralization of the tooth root resulting in a cementomimetic (e.g., cementum-like) material and, thus, can contribute to cell-based regeneration of periodontal tissue.

MATERIALS AND METHODS

Combinatorial selection of peptides
Selection of the peptides by phage display and the HAp binding and mineralization characterization were carried out as previously described 22 (see Supplementary Information).

Similarity analysis
The amino-acid sequences obtained from the combinatorially selected peptides have been used as the data source for our knowledge-based design. Using the methods previously described by Oren and colleagues, 17 two scoring matrices, HAp12I and HAp27C1, were derived for the peptides selected from the 12- and c7c-phage libraries (New England BioLabs Inc., Ipswich, MA, USA), respectively. The Point Accepted Mutation 250 (PAM 250) was chosen as the seed matrix to optimize the new matrices specific to the selected peptides. The PAM matrix gives substitution probabilities for sequences that experience a certain number of point mutations in every hundred amino acids. Therefore, PAM 250 reflects the probabilities for 250 point mutations for every 100 amino acids. PAM 250 was modified to compensate for the relative abundance of amino acids within the libraries used to select the HABP and for the codon usage of the host organism (Escherichia coli K12 ER2738) used to amplify the phages. To identify the sequence similarities between the selected peptides with the recombinant mouse 180 amino acid long amelogenin (rM180), we divided the rM180 amino-acid sequence into segments with the same length as those from the phage libraries (i.e., 7 or 12 amino acids). Each iteration generated a segment starting from the next amino acid, therefore, creating all possible 7 or 12 amino-acid segment lengths for the whole M180 sequence. Then, each segment was compared with each HABP and was given a similarity score. Once every combination of the HABPs and amelogenin segments were compared, regions that demonstrated high similarity scores against both libraries were overlapped. These coinciding high similarity regions were picked as the putative strong binding regions (see Supplementary Fig. S3). In the same way, coinciding low similarity regions were picked as the putative weak binding regions. These regions were then refined by protein structure prediction, Ca2+ ion-binding domain predictions and meta-functional signature analyses (see Supplementary Information). A schematic description of the design process is shown in Figure 1a.

Peptide synthesis
rM180 amelogenin was created as described previously. 24 The amelogenin-derived peptides (ADPs) were synthesized by standard solid phase peptide synthesis technique on Wang resin using F-moc chemistry and HBTU activation. CSBio 336s (CSBio, Menlo Park, CA, USA) automated peptide synthesizer was used for the synthesis. The resulting resin-bound peptides were cleaved and side-chain-deprotected using Reagent K (trifluoroacetic acid/thioanisole/H2O/pheno/ethanedithiol (87.5:5:5:2.5)), and precipitated by cold ether. The crude peptides obtained were purified by reverse phase high performance liquid chromatography up to a >98% purity (Gemini 10µ C18 110A column). The masses of the purified peptides were checked by mass spectroscopy using a MALDI-TOF mass spectrometer (Bruker Daltonics, Billerica, MA, USA).

Binding analysis via quartz crystal microbalance
Calcium phosphate-coated quartz crystal microbalance (QCM) electrodes were purchased commercially (Q-Sense, Västra Frölunda, Sweden). The diameter of the crystals and electrodes were 8.8 and 5.0 mm, respectively. The oscillation electronic circuit was a typical Collips oscillator, which had a buffer amplifier. A 12 V direct current was applied to the oscillator circuit to drive the crystal, and the frequency was measured with a Hewlett-Packard 53313A frequency counter sampling at 225 Hz (Universal Counter; Agilent Technologies, Santa Clara, CA, USA). After the crystal was mounted in the cell, they were cleaned and dried with high-purity nitrogen gas and used immediately. To establish a stable baseline, a sufficient amount of buffer solution was introduced into the cell before adding the peptide. The frequency change of the crystal in pure buffer solution was recorded for 30–60 min. After this step, the desired amount of ADPs was introduced into the cell and the frequency change was recorded continuously.

In vitro solution biominalization
An alkaline phosphatase based mineralization model was used as described before to investigate the mineralization behaviors of the peptides 22 (see Supporting Information for details).

Ex vivo re-mineralization of tooth root
The cementum-root stock blocks were prepared from single rooted, extracted adult teeth collected at the University of Washington Dental School clinics. No individual identifiers were used and the use of such material complied with the Institutional Review Board guidelines. The teeth were kept in 70% ethanol solution at 4°C until the specimens were prepared. Cylindrical blocks of 4 mm diameter were cut from the acellular cementum, close to the cemento-enamel junction using a trephine bur. The teeth were kept immersed in 70% ethanol-phosphate-buffered saline (PBS) during the cutting to prevent heat damage due to friction. The cut cylindrical blocks were cleaned of contaminating material using PBS and demineralized with 35% phosphoric acid gel for 10 s. The specimens were stored in 70% ethanol-PBS solution at 4°C until used.

Prior to coating the specimens with the peptide, the previously cut cylindrical blocks were rehydrated and equilibrated for 2 h in 24 mmol L⁻¹ Tris-HCl buffer, pH 7.4. The lyophilized ADP was dissolved in 24 mmol L⁻¹ Tris-HCl buffer, pH 7.4 to a final concentration of 0.4 mmol L⁻¹. A 50 µL drop of the ADP solution was dropped on top of the specimens and left on the specimen for 10 min at room temperature in a water saturated chamber to prevent evaporation. Control specimens were identically prepared only by dropping 50 µL of buffer alone. At the end of 10 min, the samples were rinsed twice with deionized water. Fluorescent microscopy analysis was performed with fluorescein isothiocyanate-labeled ADP.

Solutions of 9.6 mmol L⁻¹ CaCl₂ and 5.6 mmol L⁻¹ PO₄³⁻, a mixture of NaH₂PO₄·H₂O and Na₂HPO₄·7H₂O were prepared in 24 mmol L⁻¹ Tris-HCl buffer, pH 7.4. The peptide-coated and non-coated control specimens were placed in 300 µL of Ca²⁺ solution and an equal volume of PO₄³⁻ solution was added to achieve a final
concentration of $4.8 \text{ mmol}\cdot\text{L}^{-1}$ of $\text{Ca}^{2+}$ and $2.8 \text{ mmol}\cdot\text{L}^{-1}$ of $\text{PO}_4^{3-}$. The specimens were incubated in the mineralization solution for 2 h at 37 °C in a water saturated atmosphere, removed from the mineralization solution and rinsed with 24 mmol\cdot\text{L}^{-1} \text{Tris-HCl buffer, pH 7.4.}$

The specimens were kept in 70% ethanol solution at 4 °C until the time of characterization. A schematic explanation of the ex vivo re-mineralization is shown in Figure 2.

**Mechanical properties of the cementomimetic layer**

Mechanical properties of the cementomimetic layer were assessed by nanoindentation and qualitative mechanical abrasion tests. The samples were prepared by first infiltrating from the top of the cementomimetic layer with room-temperature-cure epoxy (Allied High Tech, Inc., Rancho Dominguez, CA, USA) to provide a continuous volume for indentation characterization. After the epoxy was cured, the sample was ground and polished to expose the interior in the cross-section of the specimen, then ultra-microtomed to achieve a smooth surface for nanoindentation. Twenty indentations each were taken using an atomic force microscope (attached with a vertical indentor) on mineralized dentin region (indentations were made on intertubular dentin), demineralized dentin regions (adjacent to the cementomimetic layer), and the cementomimetic layer (see Supplementary Fig. S7).

Qualitative mechanical analysis was carried out by ultrasonication and mechanical abrasion. For ultrasonication, specimens were
The suspended hPDL cells were prepared in serum-free DMEM and 24-well plates, with four specimens per well performed in triplicates. Hemocytometer. The equilibrated tooth specimens were placed in serum-free DMEM for 2 h. The confluent hPDL cells were suspended with 0.05% trypsin-EDTA and counted using a hemocytometer. The mineralized specimens were taken out of the 70% ethanol solution and equilibrated in serum-free DMEM for 2 h. The confluent hPDL cells were suspended in serum-free DMEM and 3×10^4 cells per well were seeded on top of the specimens. The cells were used between 2nd and 15-day durations. At the end of each time point, the specimens were taken out of the wells, rinsed with serum-free DMEM, and then with PBS. The cells were recovered from the specimen surface with 0.05% trypsin-EDTA and counted using CyQUANT cell proliferation assay kit (Life Sciences (formerly Invitrogen), Grand Island, NY, USA).

After fixing with 2% glutaraldehyde in PBS for 10 min, the cells were permeabilized with 0.1% Triton X in PBS for 2 min and blocked with 1% bovine serum albumin for 30 min and stained with Phalloidin Alexa Fluor 488 (Invitrogen) for fluorescence microscopy observations. The cells were observed and recorded using a TE 300L microscope at an appropriate wavelength and gated filters (Nikon, Tokyo, Japan).

For proliferation assays, hPDL cells were prepared the same way as described above for the cell adhesion assays. The specimens were equilibrated in media and placed in 24-well plates, four specimens per well and performed in triplicate wells. The suspended hPDL cells were prepared in serum-free DMEM and 3×10^4 cells per well were seeded on top of the specimens. After 24 h, the specimens were rinsed and transferred into new plates to prevent possible contamination of the cells that grow on the bottom of the plate. The cells were maintained in a water-jacketed incubator at 37 °C in 5% CO₂ saturated to H₂O and the media was exchanged every 24 h with DMEM supplemented with 2% fetal bovine serum. Triplicate samples were terminated after 2-, 6-, 10- and 15-day durations. At the end of each time point, the specimens were taken out of the wells, rinsed with serum-free DMEM, and then with PBS. The cells were recovered from the specimen surface with 0.05% trypsin-EDTA and counted using CyQUANT cell proliferation assay kit (Invitrogen).

**RESULTS**

The critical component in the newly developed cementomimetic mineral construction is HAp-forming peptides that operate as mineral synthesizers and control deposition of a confluent nanostructured HAp layer. The procedure for designing these peptides is schematically described in Figure 1. Using a phage display approach, we combinatorially selected more than 100 HABPs from a 7-amino acid and a 12-amino acid phage peptide library and characterized their binding affinity to the HAp mineral under near-physiological conditions. Since not all of the peptides selected have the same affinity to HAp, we next categorized them into three classes: strong-, moderate- and weak-binding peptides. Using bioinformatics classification protocols, we derived similarity-scoring matrices for both sets of selected septa- and dodeca-peptides. These matrices were used to systematically compare and identify similarity regions, i.e., domains of similar amino-acid sequences between the experimentally selected HABPs and rM180. The comparisons yielded high and low similarity regions along the amelogenin (Figure 1b). By overlapping the high-similarity regions from both libraries, putative crystal binding sequences were identified, referred to as ADPs (Figure 1b and 1e). The similarity analysis were refined and supported by other computational tools, i.e., structure prediction, meta-functional signature and ion binding domains analyses (Figure 1c and 1d) (see Supplementary Information). Many short amino-acid sequences can be generated by this procedure and each has the potential to be used for specific applications requiring control over HAp formation and growth. The putative HAp interacting regions of amelogenin having the highest similarity, i.e., ADP1, ADP2 and ADP4, were synthesized chemically (see Supplementary Information). For comparison, we also synthesized ADPs corresponding to the previously proposed putative...
Table 1 Amino-acid sequences, physicochemical properties and the dissociation constant (K_D) of the ADPs used in this study

| ADP | MW   | pl  | Charge | GRAVY  | K_D/(μmol·L⁻¹) |
|-----|------|-----|--------|--------|----------------|
| ADP1| 1414.60 | 7.10 | 0      | −0.350 | 1395           |
| ADP2| 1328.40 | 7.00 | 0      | −1.067 | 1611           |
| ADP3| 1574.70 | 4.94 | −1 (−4, +3) | −1.862 | 6397           |
| ADP4| 1833.00 | 7.16 | 0      | −0.713 | 1431           |
| ADP5| 2465.60 | 7.16 | 0 (−2, +2) | −0.959 | 50             |
| ADP6| 2630.10 | 3.79 | +2     | −0.952 | 6247           |
| ADP7| 4645.30 | 7.28 | 0      | −0.824 | 1148           |
| ADP8| 1519.80 | 5.96 | 0      | −0.743 | 314            |

ADP, amelogenin-derived peptide; GRAVY, grand average hydrophaticity; K_D, dissociation constant; MW, molecular weight; pl, isoelectric point.

Mineral-binding regions near the C-terminus (ADP3, ADP6), non-mineral interactive (e.g., low similarity score) regions (ADP5, ADP8) and finally ADP7, a peptide that included ADP1, ADP2 and ADP8 (Figure 3a and Table 1). The HAp-binding affinities of these peptides were examined via QCM and their propensity to control formation of calcium phosphates were carried out via in vitro solution mineralization assays and ex vivo tooth re-mineralization experiments.

**Binding analysis via QCM**

The dissociation constants (K_D) of the peptides were determined. The K_D is an equilibrium constant that represents the concentration necessary to achieve 50% surface coverage. Experimental binding assays demonstrated that ADP1, ADP2, ADP4 and ADP7 exhibit strong affinity to HAp with K_D values on the order of 1 μmol·L⁻¹, as was predicted by the similarity analysis (Figure 3a and Table 1). Likewise, as predicted, the binding affinities of the ADP3, ADP6, ADP5 and ADP8 were lower, with ADP5 having a significantly lower K_D (50 mmol·L⁻¹) (Figure 3a and Table 1).

**In vitro solution bio-mineralization**

We noted that there were three distinct trends of mineralization among the tested ADPs. The majority of the peptides (ADP1, ADP2, ADP3, ADP4, ADP6 and ADP8) exhibited similar kinetics to the negative control, where no peptide was present. The phage display selected HABP1 and the ADP7 identified in this study exhibited a slow mineralization trend, where more than half of the available free Ca²⁺ was consumed at the end of the 90 min (Figure 3b) (see Supplementary Fig. S5 for the mineralization trends of all ADPs).

The microstructural and crystallographic analysis of the synthesized minerals via SEM, transmission electron microscopy and X-ray diffraction (XRD) revealed another interesting consequence regarding the relationship between the mineral binding and mineralization activity. Similar to the mineralization kinetics, three distinct trends of mineral morphologies were observed. The majority of the peptides (ADP1, ADP2, ADP3, ADP4, ADP6, ADP8 and HABP1) produced spherulitic particles consistent with the formation of spherical amorphous calcium phosphate and transformation into crystalline phases. The amount of radiating crystalline blade-like particles emanating from the spherulites was slightly higher for ADP1, ADP2, ADP4 and HABP1 (strong binders) compared to ADP3, ADP6 and ADP8 (weak binders) and to the no peptide control, an order indicating that the amorphous to crystalline transformation rates were slightly different (Figure 3c). In the case of rM180 and ADP7, however, a completely different morphology was observed. Needle-like nanocrystals, organized into bundle-like assemblies, were observed for rM180 and ADP7 (Figure 3d). The bundle-like assemblies appeared to be better organized in the case of rM180, which is consistent with the self-assembly properties of amelogenin and the in vitro mineralization behavior of recombinant amelogenin, 24,32–34

The particles formed in the presence of ADP5 were much smaller spherulites with a less electron-dense core and smaller radiating crystals (Figure 3c). The smaller particles sizes in the presence of ADP5 may be due to its nucleation-dominate regime, as suggested from the data describing the mineralization kinetics (Supplementary Fig. S3).

The crystallographic analysis via XRD also confirmed that the observed morphological differences were due to different crystal structures. In all cases except rM180 and ADP7, the minerals yielded a broad peak around 2θ of 25–30° indicating a poorly crystalline phase (Figure 3c). In the presence of rM180 and ADP7, however, the XRD patterns consisted of numerous sharp peaks indicative of crystalline

![Image](image-url)
HAp. The major peaks were observed at $2\theta = 31.8^\circ (d=0.281 \text{ nm})$ and $32.1^\circ (d=0.278 \text{ nm})$ corresponding to (211) and (300) planes of HAp, respectively.

**Ex vivo re-mineralization of the tooth roots**

Human cementum discs were prepared as described in the section on ‘Materials and methods’ and as illustrated in Figure 2. Fluorescent microscopy analysis with fluorescein-labeled ADP5 showed that the ADP5 readily adsorbs on the demineralized surface of human root stock cementum and remains on the surface after extensive washing. Since the microscope was calibrated to the emission of the control samples, the background emission was eliminated in both samples. The human samples after cleaning and etching reveal that the expected dentinal tubules characteristic of the dentinocementum junction, with diameters about 1 \( \mu \text{m} \), are clearly visible on the fluorescein isothiocyanate-labeled ADP-coated samples (see Supplementary Fig. S6).

ADP5 resulted in a substantial difference in the re-mineralization profiles of the human root stock cementum surface as observed via SEM and energy-dispersive X-ray spectroscopy analysis. At the end of 2 h, no substantial re-mineralization occurred on the control samples consisting of no peptide coating (Figure 4a). On the other hand, a continuous layer of mineral covering the whole surface of the specimen was observed on the ADP5-coated specimens (Figure 4b). The morphology of the newly formed mineral was plate-like crystals growing out of the surface of the underlying dentine. Elemental composition analysis by EDS displayed no observable Ca or P peaks from the control sample containing no peptide, whereas, in contrast, substantial Ca and P peaks were observed from the ADP5-coated samples (Figure 4a and 4b inserts, respectively). Both SEM and transmission electron microscopy cross-sectional analysis showed that ADP5 yielded a 10–15 \( \mu \text{m} \) thick mineral layer that appears to be well integrated with the underlying dentin (Figure 4c and 4d). Moreover, the thickness of the cell-free biomimetic cementum layer compares well to the thickness of native human acellular cementum.

**Mechanical properties of the cementomimetic layer**

As shown in Table 2, the cementomimetic mineral layer exhibited comparable elastic modulus and hardness to the native human cementum. The large standard deviation observed for the cementomimetic layer was noted and was likely due to the less homogeneous distribution of mineral than that observed for native mineralized dentin or cementum.

We also tested whether the newly formed mineral layer can survive mechanical stresses produced by ultrasonication and mechanical abrasion (brushing). SEM observations showed that the plate-like morphology of the minerals was somewhat distorted after ultrasonication and brushing. However, as confirmed with energy-dispersive X-ray spectroscopy, the mineral was still attached to the underlying dentin (see Supplementary Fig. S8).

**Cell adhesion and growth assays**

Cell attachment assays showed that the hPDL cells attach to the mineralized surfaces more effectively compared to their non-mineralized control surfaces. Quantification of cells recovered after 2-h adhesion period showed that 23.2% of the cells were adherent on the non-mineralized control surface, whereas 60.6% of the cells adhered to the biomimetic mineralized surface after 2 h of incubation (Figure 4e).

| Layer                  | $E_r$ (GPa) | $H$ (GPa) |
|------------------------|-------------|-----------|
| Cementomimetic layer   | 19±7        | 0.7±0.5   |
| Native cementum        | 15±4        | 0.8±0.3   |
| Mineralized dentin     | 26±3        | 1.0±0.2   |
| Demineralized dentin   | 5±3         | 0.2±0.2   |

$E_r$, Young’s modulus; $H$, hardness.
Proliferation assays showed that the hPDL cells grow more efficiently on the biomimetic mineralized surfaces compared to non-mineralized control surfaces. At the end of 15 days, hPDL cells have reached more than twice the number cells on the biomimetic mineralized surfaces compared to the control surface (Figure 4f). These data suggest that the cementomimetic mineral layer provides a favorable environment for the hPDL cells to attach and to proliferate. Fluorescence imaging showed that, after initial adhesion, the cells organize their cytoskeleton on the mineral layer better (Figure 4h) than the same cells growing on control surfaces (Figure 4g).

**DISCUSSION**

The quantitative diffraction and microscopy analyses of the minerals produced by the ADPs revealed that, among the designed peptides, only ADP7, the longest of the similarity regions, and native rM180 produced needle-like, crystalline HAp particles, while the others formed crystalline spherulite particles (Figure 3c–3e). Interestingly, the N-terminal ADP5, which revealed no binding affinity to HAp, exerted a kinetic control over calcium phosphate nucleation at a level similar to full-length rM180 amelogenin protein (Figure 3b). The fact that the ADP5 did not demonstrate a high affinity to HAp indicates that control over nucleation is governed by interactions with precursor ions. As suggested previously in the literature, these data indicate that the amelogenin protein may be divisible into individual domains that exert separate control over calcium phosphate crystal nucleation, growth and morphogenesis.36–40

Taken together, these data indicate that the strong binding affinity by a peptide to its mineral substrate does not necessarily always translate into mineralization directing activity. The fact that the ADP5 showed no significant binding affinity towards HAp but resulted in faster kinetics implies that the ADP5 might be interacting with the soluble precursor ions rather than the mineral surface. The two adjacent pairs of oppositely charged residues at positions 9–10 and 19–20 within ADP5 may be responsible for attracting soluble Ca$^{2+}$ and PO$_4^{3-}$ and creating an increased local supersaturation and, thereby, decreasing the nucleation barrier. In contrast, the strong interaction between ADP7 and the mineral surface indicates that a change in the mineral–solution interfacial energy is responsible for the observed mineralization behavior. It has been reported that biomolecules, such as citrate, can alter the interfacial energy of kinetically favorable metastable calcium phosphate phases and trigger the phase transformation towards thermodynamically favorable HAp.41–42 A similar effect may be responsible, due to the relatively high affinity of ADP5 to the mineral surface. Binding of a peptide on a crystal surface is determined by both the conformation (side chain availability) of the peptide and the atomic configuration of the crystal at the binding interface.43–45 We speculate that the formation of HAp instead of other metastable CaP phases, such as amorphous calcium phosphate, octacalcium phosphate or tricalcium phosphate, in the presence of ADP7, is due to a specific conformation of the peptide and the arrangement of Ca and P sites on the crystal surface that maximizes the binding energy and, in turn, reduces the interfacial energy barrier required to overcome phase transformation.

The binding and biomineralization behaviors of ADP5, ADP7 and rM180 imply an interesting mechanism for amelogenin-mediated mineralization during the formation of enamel. The fact that two different ADPs emulate different aspects of the mother protein brings up the question whether these peptides might be different functional regions within amelogenin.37 Namely, ADP5 may be the region responsible for increased mineralization kinetics through increasing local supersaturation and ADP7 may be the region responsible for favoring the phase transformation towards HAp through altering the interfacial energy. However, the answers to these questions are the focus of another ongoing study.

The range of characteristics exhibited by these individual ADPs represents potential molecular agents to consider for engineering and biomedical applications involving hard tissue repair and replacement where specific control features for the mineral phase are desired. Peptide-assisted re-mineralization of dental tissue defects offers a significant potential for clinical use. Towards this end, we asked whether or not the biomimetic cementum-like mineral layer has a favorable effect on cell behavior *in vitro*. Cultured hPDL were selected for investigation since these are the predominant cell types that root surfaces are exposed to *in vivo*, and further, these cells, when appropriately triggered, can carry on important functions in the repair and regeneration of periodontal tissues.46–50 Cell adhesion and cell proliferation assays, performed on the surface of ADP5-mediated cementum-like tooth material in the absence of serum-derived factors, showed that the nanostructured mineral layer favors both the adhesion and proliferation of cells derived from the periodontium (Figure 4e and 4f). We are presently refining conditions for generating the cementomimetic layer seeking to engineer further improvements to attachment and proliferation so as to improve the regenerative competency of this material.

In conclusion, the work described here has several significant implications in peptide-assisted biomineralization. First, we describe a new protocol for identifying peptide sequences from among native proteins associated with HAp-containing tissues that are critical to their control over biomineralization. A similar analysis of extracellular matrix proteins associated with biominerals formed by unicellular organisms (e.g., magnetotactic bacteria), invertebrate (sponges, molusks) and vertebrate animals can reveal other unique amino-acid domains that regulate the mineral formation and growth for each of their variety of inorganic compounds (magnetite, silica and calcium carbonate polymorphs).51 Biomineralization is fundamental to many living organisms and the range of precipitated minerals reflects the elements of the periodic table. Elaborating the mechanism for protein control over the biominions formed in these systems will afford new insights into the evolution of protein–mineral interactions.52 In the case of enamel, amelogenin is a member of the class of intrinsically disorganized proteins.53 By restricting a structural analysis to a small domain of a larger protein, such as that defined by ADPS, it is possible to expand our understanding of structure–function relationships, portending the capacity to decipher a relationship between a peptide and a chemical precipitate, and then to use such information for their practical utility to engineer material formation technologies (e.g., bionanomaneufacturing). Secondly, as demonstrated in this study, strong binding affinity of a peptide to an inorganic solid is not necessarily an indication of its mineral formation capability. Thirdly, the cementomimetic layer formed by ADP5 has a clinical application potential to repair diseased root surfaces, both those caused by caries and periodontal disease, and to promote regeneration of periodontal tissue, reducing the morbidity associated with tooth loss.

**AUTHOR CONTRIBUTIONS**

MG, MH and CT identified the combinatorial selected peptides; MG carried out the binding studies with QCM, carried out XRD and cell culture experiments and contributed to the content of the manuscript; MG and HF carried out mineralization and microscopy studies; M
Sarikaya, EEO, JAH and RS designed the amelogenin derived peptide sequences. EEO, JAH, and RS performed bioinformatics and computational modeling of amelogenin; RS, JAH and EEO identified meta-functional signatures and ion-binding domains; MLS provided mouse amelogenin, contributed to the dissection of amelogenin domains and functions and contributed to the content of the manuscript; CT supervised MG, HF and MH, contributed to the creation of genetically engineered peptides and contributed to the content of the manuscript; MJS provided the cells, directed the research toward cementum regeneration and contributed to the content of the manuscript; M Sarikaya proposed to investigate similarity regions along amelogenin using combinatorial library-selected peptide sequences towards HAp bio-mineralization, directed the research, and contributed to the content of the manuscript.

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1 Lowenthal MA, Weiner S. On biomineralization. New York: Oxford University Press, 1989.
2 Mann S. Biomineralization: principles and concepts in biomineralizing materials. New York: Oxford University Press, 2001.
3 Begue-Kirn C, Krebsbach PH, Bartlett JD et al. Materials Research Facilities Network. The authors thank Dr B R Wilson for
4 3540. Material self-assembly creates a nanoscale device for biomineralization. Mater Sci Eng C 2006; 26(8): 1296–1300.
5 Du C, Falini G, Farnesi S et al. Supramolecular assembly of amelogenin nanospheres into birefringent microspheres. Science 2005; 307(5714): 1450–1454.
6 Gungormus M, Gungormus C, Krebsbach PH, Bartlett JD et al. Materials Research Facilities Network. The authors thank Dr B R Wilson for
7 74: 57–115.
8 Iijima M, Moradian-Oldak J. Interactions of amelogenins with octacalcium phosphate crystal faces are dose dependent. Calcif Tissue Int 2004; 74(6): 522–531.
9 Fan D, Iijima M, Bromley KM et al. The cooperation of enamelin and amelogenin in controlling octacalcium phosphate crystal morphology. Cells Tissues Organs 2011; 194(2/3/4): 194–198.
10 Gungormus M, Fong H, Kim IW et al. Regulation of in vitro calcium phosphate mineralization by combinatorially selected hydroxyapatite-binding peptides. Biomacromolecules 2008; 9(3): 966–973.
11 Moradian-Oldak J, Paine ML, Lei YP et al. Self-assembly properties of recombinant engineered amelogenin proteins analyzed by dynamic light scattering and atomic force microscopy. J Struct Biol 2000; 131(1): 27–37.
12 Pugach MK, Li Y, Sughs C et al. The amelogenin C-terminus is required for enamel development. J Dent Res 2010; 89(2): 165–169.
13 Friddle RW, Battle K, Trubetskoy V et al. Single-molecule determination of the face-specific adsorption of amelogenin’s C-terminus on hydroxyapatite. Angew Chem Int Ed 2011; 50(33): 7541–7545.
14 Shaw WJ, Campbell AA, Paine ML et al. The CDOH terminus of the amelogenin, LRAP, is oriented next to the hydroxyapatite surface. J Biol Chem 2004; 279(39): 40263–40266.
15 Aoba T, Moreno EC, Kresak M et al. Possible roles of partial sequences at N- and C-termini of amelogenin in protein-enamel mineral interaction. J Dent Res 1989; 68(9): 1311–1336.
16 Arya A, Jedwab J, Pireaux J et al. Brushite in the pulp of primary molars. J Oral Pathol Med 1999; 18(7): 371–376.
17 Kodaka T, Hirasaya A, Mori R et al. Spherulitic brushite stones in the dental pulp of a cow. J Electron Microsc 1998; 47(1): 57–65.
18 Achilles W, Jockel U, Schaper A et al. In-vitro formation of urinary stones—generation of spherulites of calcium-phosphate in gel and overgrowth with calcium-oxalate using a new flow model of crystallization. Scanning Micr 1995; 9(2): 577–586.
19 Margolis HC, Beniash E, Fowler CE. Role of macromolecular assembly of amelogenin in protein matrix enamel in formation. J Dent Res 2006; 85(9): 775–793.
20 Beniash E, Simmer JP, Margolis HC. The effect of recombinant mouse amelogenin on the formation and organization of hydroxyapatite crystals in vitro. J Struct Biol 2005; 149(2): 182–190.
21 Fan Y, Sun Z, Moradian-Oldak J. Controlled remineralization of enamel in the presence of amelogenin and fluoride. Biomaterials 2009; 30(4): 478–483.
22 Ho SP, Yu B, Yun W et al. Structure, chemical composition and mechanical properties of human and rat cementum and its interface with root dentin. Acta Biomater 2009; 5(7): 707–718.
23 Moradian-Oldak J, Paine ML, Lei YP et al. Carboxy- and amino-terminal domains of amelogenin are involved in the supramolecular self-assembly. J Dent Res 2000; 79: 513–513.
24 Moradian-Oldak J, Paine ML, Luo W, Zhu DH et al. Functional domains for amelogenin revealed by compound genetic defects. J Bone Miner Res 2003; 18(3): 466–472.
25 Sneed ML. Amelogenin protein exhibits a modular design: implications for form and function. Connect Tissue Res 2003; 44: 47–51.
26 Dunglas C, Septer D, Paine ML et al. Ultrastructure of forming enamel in mouse bearing a transgene that disrupt amelogenin assembly domains. J Dent Res 2001; 80(4): 1278–1278.
27 Le Norcy E, Kwak SW, Wiedemann-Bidlack FB et al. Potential role of the amelogenin N-terminus in the regulation of calcium phosphate formation in vitro. Cels Tissues Organs 2011; 194(2/3/4): 188–193.
28 Qiu SR, Wierzbicki A, Orme CA et al. Molecular modulation of calcium oxalate crystallization by osteopontin and citrate. Proc Natl Acad Sci U S A 2004; 101(7): 1811–1815.
29 Jiang WG, Chu XB, Wang B et al. Biomimetically triggered inorganic crystal transformation by a molecular self-assembly: a new understanding of biomineralization. J Phys Chem B 2009; 113(31): 10838–10844.
30 Elangovan S, Margolis HC, Openheim FG et al. Conformational changes in salivary proline-rich protein I upon adsorption to calcium phosphate crystals. Langmuir 2003; 19(22): 11200–11207.
31 Masica DL, Gray JJ. Solution- and adsorbed-state structural ensembles predicted for the statherin-hydroxyapatite system. Biophys J 2009; 96(8): 3082–3091.
32 So CR, Tamerlik C, Sankiaya M. Adsorption, diffusion, and self-assembly of an engineered globular (pH) peptide on Au(111) investigated by atomic force microscopy. Angew Chem Int Ed 2009; 48(28): 5174–5177.
33 Nagatomo K, Komaki M, Sekiya I et al. Stem cell properties of human periodontal ligament cells. J Periodont Res 2006; 41(4): 303–310.
34 Zeng Y, Miura M, Gultunova O et al. Investigation of multipotent postnatal stem cells from human periodontal ligament. Lancet 2004; 364(9429): 149–155.
35 Zeng Y, Miura M, Sonomaya W et al. Recovery of stem cells from cryopreserved periodontal ligament. J Dent Res 2005; 84(10): 907–912.

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49 Gay IC, Chen S, MacDougall M. Isolation and characterization of multipotent human periodontal ligament stem cells. Orthod Craniofac Res 2007; 10(3): 149–160.

50 Ivanovski S, Gronthos S, Shi S et al. Stem cells in the periodontal ligament. Oral Diseases 2006; 12(4): 358–363.

51 Sarikaya M, Aksay IA. Biomimetics: design and processing of materials. Woodbury: NYAIP Press, 1995.

52 Laksminarayanan R, Bromley KM, Lei YP et al. Perturbed amelogenin secondary structure leads to uncontrolled aggregation in amelogenesis imperfecta mutant proteins. J Biol Chem 2010; 285(52): 40593–40603.

53 Delak K, Harcup C, Lakshminarayanan R et al. The tooth enamel protein, porcine amelogenin, is an intrinsically disordered protein with an extended molecular configuration in the monomeric form. Biochemistry 2009; 48(10): 2272–2281.

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