Graphite-like dynamical behaviour of graphite oxide

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Abstract. We report two-colour pump-probe spectroscopy of coupled structural and electronic dynamics of graphite oxide probed with 6-µm and THz pulses upon femtosecond IR excitation.

1 Introduction

Since the discovery of graphene’s unique electronic structure with linear dispersion of Dirac electrons graphite oxide (GO) has gained attention due to its potential in mass production of graphene. However, it is increasingly apparent that GO is an interesting material of its own with various applications from plastic electronics, optical materials and catalysis to biosensors. Its heterogeneous chemical and electronic structures engender from a variety of oxygen functional groups which have been identified as hydroxyl (OH), epoxy (-O-), carbonyl (C=O), and carboxylic (-COOH) groups alongside five- and six-membered-ring lactols. These groups are attached to the two-dimensional graphene layer, thereby distorting the planar geometry of the graphene sheets [1]. The functionalization of graphene thus leads to non-stoichiometric macromolecules with a complicated structure. The particular distribution of these functional groups will determine domain structure and local band gaps as well as carrier dynamics and energy transfer mechanisms. Due to the oxygen-containing functional groups GO is highly hygroscopic. As a consequence, water is trapped within the ~1 nm spacing between individual graphene oxide sheets [2], leading to additional energy relaxation channels.

We report on two-colour pump-probe spectroscopy to study vibrational/phonon and carrier dynamics after excitation of the ~14 µm thick self-supporting GO membrane with femtosecond pulses from the UV to the mid-infrared (MIR) and probed the material response in the MIR and THz region. Probing in the MIR spectral region gives access to vibrational (phonon) dynamics, vibrational (phonon-phonon) coupling and vibronic (electron-phonon) coupling between electronic and nuclear degrees of freedom. The THz spectral region complements the insight into the materials response to yield information on charge carrier dynamics as they exhibit a distinct optically active response in the MIR and THz frequency range.

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2 Experimental and results

Ultrafast responses of GO in the MIR region upon 2-µm (620 meV) excitation are shown in Fig. 1B with static absorption (top) and time-resolved differential absorption spectra (bottom). The MIR spectral region between 1500 and 1900 cm\(^{-1}\) is resonant with the in-plane stretching modes (phonons) of the sp\(^2\)-hybridized C=C and carbonyl (C=O) groups. Upon excitation at 2 µm the transient MIR spectra show two distinct features superimposed on a broad transient background (characteristic of free charge carriers). These two spectral features and the electronic background appear within the experimental time resolution of 150 fs and subsequently decay with very similar kinetics. The transient inverted band around 1570 cm\(^{-1}\) can be assigned to a reduced C=C stretching response. The second inverted band around 1673 cm\(^{-1}\) can be assigned to the in-plane carbonyl stretching vibration of an \(\alpha,\beta\)-unsaturated carbonyl, the so called enone (Fig. 1A illustrated the enone functional group within a GO sheet).

![Fig. 1. A Schematic GO model with enone (blue), epoxy (red) and hydroxyl (green) groups. B Static and transient spectra of GO (\(\Delta t\): 0.15ps to 2ps). C Time traces of the \(\nu_{C=C}\), \(\nu_{C=O}\), and carrier response. D THz response of GO after 2µm excitation.](https://doi.org/10.1051/epjconf/201920504014)

In order to quantitatively retrieve the decay time constants for each signal contribution a line-shape fitting procedure has been used to model the two transient bands with asymmetric Gaussians and the broad electronic background. The evolution of selected structural line-shape parameters and the electronic background (offset) are displayed in Fig. 1C. A modified two-temperature model with the shortest two time constants of 150 fs and 600 fs and an intermediate time constant of about 4 ps can be fit to the area of both transient bands and the broad underlying signal independent of the pump wavelength. The frequency shift of the C=C stretching vibration occurs with a time constant of about 600 fs. A small long-time component extends to hundreds of picoseconds.

The first time constant can be assigned to carrier-phonon scattering. For graphite photoexcited carriers transfer most of their energy to a set of strongly coupled optical phonons (SCOPs), including the C=C ring stretching vibration (G-mode phonon) within less than 300 fs [3] leading to a significant non-equilibrium phonon population. Subsequent phonon-phonon scattering proceeds with a time constant of 2.2 ps. The cooling process manifests itself in a relaxation of the G-phonon frequency shift of about 10 cm\(^{-1}\) upon carrier relaxation due to electron-phonon coupling increasing to that of the unexcited sample, reminiscent of dynamical behaviour in graphene and graphite. [3] Compared to graphite the 600 fs time constant points to a faster cooling process in GO. Possibly due to the higher vibrational density of states in GO compared to graphite in part provided by the oxygen-containing functional groups and intercalated water. The observation that the in-plane
carbonyl stretching vibration shows the very same dynamics as the C=C stretching vibration hints at strong coupling between these two modes. Indeed, a recent two-dimensional infrared spectroscopic study of a series of enone-containing quinones reports significant vibrational coupling between these two modes [4].

GO studies employing Raman spectroscopy, scanning tunnelling microscopy and high-resolution transmission electron microscopy report irregular coverage of oxygen-containing functional groups on the graphene sheets [1]. Ordered small sp² carbon domains exist within the sp³ C-O matrix. Calculations have shown that such isolated molecular sp² domains within the sp³ matrix can lead to confinement of π-electrons in GO [5]. The smaller these sp² domains, the higher the local energy gap due to increased spatial confinement. Conversely, the large bandgap distribution manifesting in absorption from meV to eV reflect a large range of sp² domain sizes in GO without a distinct band gap. Our experimental findings show that photo-generation of electron-hole pairs is feasible even at photon energies of ~200 meV and below. This implies the existence of sp² domains substantially larger than 2nm but other sp² configurations such as different shapes and topologies and in-plane edge carbonyl groups present in GO should be considered as well.

The generation of charge carriers in GO induced by optical excitation at 2 µm has also been probed as a change in the THz peak electric field. The resulting kinetic trace is depicted in Fig. 1D. Besides the carrier decay dynamics the THz probe pulses may also induce dynamics in the sample as well. For a correct description of the THz response, we use a third-order approach which includes finite dephasing times for the THz-induced polarization and proper convolution of response functions. By doing so, the elevation at positive times can be correctly modelled, also by using the time constants gained from IR probe measurements. The signals at negative time delays may be interpreted as perturbed free induction decay which would explain the discrepancy between pump pulse duration and the EOS response function on one hand and the apparently larger THz response of GO with recurrences reminiscent of the field envelope of the THz probe pulse on the other.

Interpretation of such kind of decay is under current discussion in the literature which leads us to mention that electron-electron and electron-phonon scattering play a key role in the decay mechanism during the first picoseconds [6] but the time constants for these are not clearly attributed in the literature, yet [7]. We suspect that the last decay step is via trapping of carriers in mid-gap states arising from numerous defects leading to relaxation timescales of tens of picoseconds. Small persistent signals on timescales of hundreds of picoseconds point to occupied defect states and diffusive carrier motion with increasingly scarce unoccupied trapping states.

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