Metal Surface Modification for Obtaining Nano- and Sub-Nanostructured Protective Layers

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Abstract

Regularities of the phase protective layer formation in multicomponent systems involving inhibitors with different mechanism of protective action have been investigated. It was shown that optimization of the composition of the inhibition mixture allows to obtain higher protective efficiency owing to improved microstructure of the phase layer. It was found that mechanism of the film formation in the presence of NaNO2-PHMG is due to deposition of slightly soluble PHMG-Fe complexes on the metal surface. On the basis of the proposed mechanism, the advanced surface engineering methods for obtaining nanoscaled and sub-nanostructured functional coatings may be developed.

Keywords: Metal protection, Surface engineering, Pitting corrosion, Isomolar series, Complexes, Phase layers

Background

Protection of metals under different conditions in various environments remains the important issue for science and technology. Recent advances in research of the nanoscaled systems allow the development of simple and inexpensive methods for obtaining nanostructured protective coatings with improved functional properties.

The protection mechanism of certain types of corrosion inhibitors is based on the formation of the phase protective layers on the metal surface. Among them, one may list the inorganic passivators that facilitate the formation of the dense oxide or salt protective films and complexing type inhibitors that are forming phase metalorganic layers. The mechanism of protective activity of the complexing type inhibitors consists in the formation of slightly soluble complexes of organic ligands with the metal ions. Deposition of such compounds on the metal surface in form of the phase layers leads to the gradual decrease in corrosion rate thus providing the efficient anticorrosive protection. The crucial parameters for ensuring high protective efficiency of the complexing type inhibitors are the high stability and low solubility of complexes with the cations of the metal being protected.

Establishing regularities of the phase layer formation processes in multicomponent systems on a nano- and sub-nanoscale will enable the development of novel surface engineering techniques for obtaining functional coatings with desired functional properties.

The present study aimed to investigate the joint action of the inhibitors with different protective mechanism as well as their influence on the phase layer microstructure and achieved anticorrosive efficiency.

Methods

The studied materials were as follows: steels 08kp and 20, corrosion environment—aqueous saline solutions with compositions of 0.3 g/l NaCl, 0.3 g/l Na2SO4, and 0.3 g/l NaHCO3, inhibitors—NaNO2, Na2SiO3, polyhexamethylene guanidine (PHMG), and their isomolar mixtures (with total concentration of 30 mmol/l).

Corrosion tests and determination of the inhibitor efficiency were performed gravimetrically according to the standard procedure. For this purpose, specimens were exposed into corrosion environment for different times from 72 to 168 h. The ratio of solution volume
(ml) to the area of metal sample (cm²) was a 10:1. The inhibition efficiency was determined by the equation

\[ Z = \frac{K_m - K_{0m}}{K_m} \times 100\% \]

where \( K_m \) and \( K_{0m} \) are the corrosion rates in solutions without and with inhibitor correspondingly (\( K_m = \frac{\Delta m}{S \cdot \tau} \)), where \( \Delta m \) is the loss of the sample weight, g; \( S \) is the sample area, m²; \( \tau \) is exposure time, h).

Polarization measurements were carried out in potentiostatic regime in a three-electrode cell with separated cathodic and anodic compartments. Carbon steel was used as the working electrode, platinum—as the counter-electrode and an Ag|AgCl|KCl (sat.) electrode—as the reference one. In this paper, the potential values are given with respect to the normal hydrogen electrode potential.

The morphological characteristics and elemental composition of the films were carried out with the scanning electron microscope (EVO-50, Zeiss, Germany) equipped with the energy-dispersive detector (INCA PENTA FET x 3, Oxford Instruments, Co., UK). Elemental analysis of the protective thin films has been carried out using Auger microprobe JAMP-9500F in the scanning electron microscopy mode.

Investigation of the complex formation process and geometry optimization has been carried out with a hybrid QM/MM approach using NWChem 6.5 computational chemistry package [8].

**Results and Discussion**

The results of gravimetric tests show high protective efficiency of NaNO₂ in aqueous saline solutions. The inhibition efficiency reaches a value of 99.2% at the sodium nitrite concentration of 30 mmol/l. The polarization measurements in the presence of 30 mmol/l NaNO₂ show the potential shift toward the positive side that leads to increased corrosion resistance of the carbon steel. The passive zone extends on a wide range of potentials with significant drop in current density (Fig. 1).

The protection mechanism of NaNO₂ usually attributed to its ability to facilitate the formation of dense Fe₃O₄ or mixed oxide films that prevent further corrosion damage [9]. Such properties may be associated with the electronic structure of NO₂⁻ ion due to the presence of lone pair of electrons in sp²-hybrid orbital that provides with the electron-donor properties in contrast to NO₃⁻ ions that do not exhibit protective properties (Fig. 2).

SEM images of the samples immersed into the NaNO₂ solution reveal the significant changes in the morphology attributed to the formation of protective film as well as signs of corrosion attack (Fig. 3 c, d). While the gravimetric measurements have shown high inhibitive efficiency, the surface morphology has significant amount of localized corrosion damage. The Auger...
spectroscopy has been employed to investigate the elemental composition of the protective layer. Obtained results indicate presence of the iron oxide and are found to be in a good agreement with theoretical expectations. Local types of corrosion such as pitting are considered as the most dangerous ones; hence, this drawback must be addressed during development of the inhibitive compositions based on sodium nitrite.

To achieve the synergistic effect, the inhibitive composition should combine the components with different mechanism of the anticorrosive action [10–13]. The inhibition mechanism of Na$_2$SiO$_3$ is attributed to a formation of slightly soluble silicates of iron on the metal surface. The distribution of the protective efficiency within the NaNO$_2$-Na$_2$SiO$_3$ inhibitive mixture in relation to the concentration ratio of the components has been obtained using the isomolar series method (Fig. 4).
The regularity shows extreme character reaching peak value of \( z = 99.97\% \) at the concentration ratio \( \text{NaNO}_2-\text{Na}_2\text{SiO}_3 \) as 1:2, which is higher than protective efficiency of both individual components.

Polarization measurements for this mixture are found to be in a good agreement with the gravimetric studies (Fig. 1). The inhibitive composition of \( \text{NaNO}_2-\text{Na}_2\text{SiO}_3 \) ensures significant drop in the current density in a wide range of potentials that correspond to a passive state of the surface. SEM images reveal significantly improved morphology of the protective film formed in presence of the \( \text{NaNO}_2-\text{Na}_2\text{SiO}_3 \) mixture compared to sole sodium nitrite solution (Fig. 4f). Obtained protective film has continuous structure without signs of pitting corrosion. Elemental composition of the protective layer showed presence of the silicon that indicates direct participation of \( \text{SiO}_3^{2-} \) ions in the film formation process. While the \( \text{NaNO}_2-\text{Na}_2\text{SiO}_3 \) mixture showed significant improvement in inhibition coefficient over sole sodium nitrite, the most valuable outcome of the composition optimization is greatly enhanced efficiency toward local corrosion damage.

To ensure reliable corrosion protection for diverse applications, the inhibitive composition should provide efficient reduction in corrosion rate (in wide range of pH and) under elevated temperature conditions. At the same time, it was found that \( \text{NaNO}_2-\text{Na}_2\text{SiO}_3 \) mixture demonstrates high protective efficiency at room temperature, while the protection coefficient declines as the temperature rises.

Further improvement in protective performance may be achieved by combining inorganic passivator with organic ligands that could form complexes with the metal ions. The mechanism of protective activity of complexing type inhibitors for acidic medium has been proposed and extensively studied over the last decade [6, 7]. In such systems, protective layers are formed as a result of formation of the slightly soluble complexes of added organic ligands and the metal cations. Deposition of such compounds leads to the formation of the phase metalorganic layers on the metal surface and subsequent decrease in corrosion rate. Thus, application of complexing type inhibitors allows one to implement in situ surface modification for obtaining hybrid protective and functional coatings.
Polyhexamethylene guanidine (PHMG) showed high inhibitive efficiency for protection of carbon steel in acidic medium [6, 7, 13, 14] while exhibiting moderate performance in neutral aqueous saline solutions. Gravimetric measurements showed that PHMG provides 40% corrosion protection at the concentration of 30 mmol/l.

Relatively low efficiency of sole PHMG may be attributed to the fact that according to the protection mechanism of complexing type inhibitors, the corrosion process should provide sufficient amount of Fe$^{2+}$ ions for the complexation. Suchlike state is spontaneously established and maintained in acidic medium, while in neutral aqueous saline solutions, such conditions should be engineered according the Pourbaix diagram for the particular system. Hence, sodium nitrite can stimulate the localized pitting attack; it may indirectly facilitate to establishing of the favorable conditions and engage the complexing type component of the inhibitive mixture. In addition, PHMG protection performance tends to increase under elevated temperature conditions that being combined with NaNO$_2$ may further improve the efficiency of such mixture.

The isomolar series method showed a sharp increase in the protective efficiency with addition of sodium nitrite while the peak value of 99.8% is reached at the concentration ratio NaNO$_2$-PHMG as 2:1 (Fig. 5). Polarization measurements are found to be in a good agreement with gravimetical tests. Analysis of the polarization curves showed that electrochemical behavior of the NaNO$_2$-PHMG mixture is close to sole NaNO$_2$ with extended passive zone in a wide range of potentials at the early stages of the measurements, while acting similar to PHMG at the later stages.

Investigation of the surface morphology after immersion in the solution with PHMG (Fig. 6a) expectedly shows significant amount of corrosion products mixed with organic inclusions that is also consistent with the Auger spectroscopy results (Fig. 7a, Table 1). Whereas in the case of NaNO$_2$-PHMG mixture, the surface morphology is in almost pristine condition (Fig. 6b) while the elemental composition of the surface layer indicates presence of the protective film of mainly organic nature (Fig. 7b, Table 1).

Further analysis of the Auger spectra indicates the formation of nanoscaled protective film as a result of tandem action of the mixture components where NaNO$_2$ facilitates establishing the conditions for the complexation process and formation of metalorganic layers with PHMG-Fe complexes. According to the hybrid QM/MM calculations [14], the PGMG-Fe$^{2+}$ complexes geometry will contribute to the formation of irregular three-dimensional mesh frameworks that will determine the sub-nanoscaled structure of the phase metalorganic layers.

Conclusions
Regularities and mechanism of the protective layers formation in presence of synergistic mixtures of the corrosion inhibitors with different mechanism of action have been investigated. The isomolar series method has been employed for determination of optimal concentration ratios of the mixture components. For the NaNO$_2$-Na$_2$SiO$_3$ mixture, the maximum protective efficiency of $z = 99.97\%$ is achieved at the concentration ratio as 1:2, while for NaNO$_2$-PHMG system $z = 99.8\%$ at the concentration ratio as 2:1.

It was found that NaNO$_2$-Na$_2$SiO$_3$ mixture demonstrates much improved anticorrosive behavior compared to sole sodium nitrite due to suppressing the pitting corrosion. The NaNO$_2$-PHMG mixture ensures the formation of thin metalorganic layers with PHMG-Fe complexes and provides better protection under elevated temperature conditions.

The mechanism of joint action of the NaNO$_2$-PHMG inhibitive mixture for protection of carbon steel in aqueous saline medium has been proposed. Sodium nitrite may indirectly facilitate to the establishing of the favorable conditions for the complexation process and engage the complexing type component of the mixture.

The inhibitive mixtures with such design where one component tuned to promote the most efficient realization of the other component protective mechanism may be distinguished to a dedicated class of tandem inhibitors.

On the basis of the proposed mechanism, the advanced surface engineering methods for obtaining nanoscaled and sub-nanostructured functional coatings may be developed.

Authors’ Contributions
VL developed the plan of the study for inorganic inhibitive mixtures and took part in the general analysis of the result and preparation of the article. YuV and IB developed the plan of the study for organic inhibitive mixtures, carried out the electrochemical investigation, and took part in the general analysis of the result and preparation of the article. VL carried out the corrosion rate study. All authors read and approved the final manuscript.

Competing Interests
The authors declare that they have no competing interests.

Table 1 Key to Fig. 7: elemental composition of the protective films with Auger

| Figure | C   | N   | O   | Na  | Si  | S   | Cl  | Fe  |
|--------|-----|-----|-----|-----|-----|-----|-----|-----|
| 7a     | 12.3| 5.5 | 55.6| 0.0 | 0.1 | 0.1 | 0.1 | 26.4|
| 7b     | 48.1| 16.1| 22.3| 0.5 | 1.8 | 0.3 | 0.1 | 10.9|

All values are in atomic %
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