Sn Quantum Dots for Electrocatalytic Reduction of CO$_2$ to HCOOH

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Abstract: Sn based materials, as low-cost and earth-abundant electrocatalysts, are potential candidates for CO$_2$ reduction reaction (CO$_2$RR) into liquid fuels. Unfortunately, the low selectivity and stability limits their applications. Herein, we developed an electrocatalyst of Sn quantum dots (Sn-QDs) for efficient, durable and highly selective CO$_2$ reduction to HCOOH. The Sn-QDs were confirmed with high crystallinity and an average size of only 2-3 nm. The small particle size endowed the electrocatalyst with improved electrochemical active surface area (ECSA), which was about 4.4 times of that of Sn particle. The enlarged ECSA as well as accelerated CO$_2$RR kinetics favored the electrochemical conversion of CO$_2$. The Faradaic efficiency of HCOOH (FE$_{HCOOH}$) on Sn-QDs/CN reached up to 95% at -1.0 V (vs RHE), which exceeded 83% in the recorded wide potential window of 0.5 V. Moreover, the Sn-QDs electrocatalyst exhibited good electrochemical durability for 24 h.

Key words: CO$_2$ reduction; Sn quantum dots; HCOOH; electrocatalyst

With the increase of atmospheric CO$_2$ concentration, many eco-environmental problems are increasingly serious. Global endeavors are taken to capture and store or convert/utilize CO$_2$, in order to achieve zero carbon emission in future$^{[1]}$. Electrocatalytic CO$_2$ reduction reaction (CO$_2$RR) receives extensive concern, because it can be operated under ambient pressure/temperature and is possible for scale-up industrialization$^{[3]}$. CO$_2$ molecules can be converted into highly value-added carbon-containing products by sustainable electricity$^{[3]}$. However, multiple proton and electronic transfer steps involved in CO$_2$RR make the reaction kinetically sluggish, resulting in high overpotential or low Faradaic efficiency$^{[1,4]}$. Therefore, it is urgent to develop highly efficient electrocatalysts to reduce CO$_2$ into target products with high selectivity at low overpotentials.

The pathways of CO$_2$RR involve successive CO$_2$ adsorption and activation, intermediate formation and conversion, finally products desorption from electrocatalyst surface$^{[5]}$. Previous publications$^{[6-11]}$ elucidated that the catalytic activity and production selectivity of metallic catalysts have a striking relation with their particle size, as the electronic structure and the adsorption of intermediates always vary according to particle size. Generally, with the size decrease of metallic catalyst, the electrochemical active surface area (ECSA) and the number of active sites on its surface for CO$_2$RR increases. Likewise, in a certain size range, the activity of metallic catalyst and the Faradaic efficiency (FE) of products increase as the size of metallic catalyst decreases$^{[6,12]}$. For example, Gu’s group$^{[11]}$ found out variance in size effect of Ni metal on CO$_2$RR performance. Ni single atoms exhibited superior CO$_2$RR activity with FE$_{CO}$ of 97%. While, for Ni particles, the FE$_{CO}$ slightly lowered to 93% at the particle size of 4.1 nm and even decreased to below 30% accompanied with FE$_{H2}$ of above 70% at the particle size of 37.2 nm.
Quantum dots (QDs) are desirable electrocatalysts for CO$_2$RR owing to their extra-small size, high dispersity and unique electronic properties$^{[13-15]}$. For instance, N-doped graphene quantum dots (NG-QDs) achieved high CO$_2$ electro-reduction activity with high current density at low overpotentials. The increased pyridinic N species at edges were recognized as active sites, and the decreased size of NG-QDs brought fastened kinetics$^{[15]}$. Metal QDs catalysts (Pb, Au, Ag QDs as well as Cu QDs) derived from metal chalcogenide QDs exhibited drastically enhanced CO$_2$RR performances partially resulting from the increased propensity of CO$_2$ adsorption onto their surface$^{[14]}$. The experiments and DFT calculations demonstrated that maximizing the atomic defects density in QDs favoured the localization of atomic and electronic structure, and lowered the energy barrier of CO$_2$RR. Inspired by these results, it is rational to design metallic catalysts with small size for improved CO$_2$RR performance. Herein, we synthesized Sn-QDs for CO$_2$RR to produce HCOOH via an in-situ electrochemical reduction process.

1 Experimental section

Sn particle decorated polymeric carbon nitride, denoted as Sn-p/CN, was prepared according to our previous work$^{[16]}$. Sn quantum dots decorated polymeric carbon nitride, named as Sn-QDs/CN, was obtained via an in-situ electrochemical reduction process on Sn-p/CN in a three-electrode system. Typically, 20.0 mg of the as-prepared Sn-p/CN was dispersed in 2 mL of isopropanol and 80 μL of Nafion solution (5wt%) under sonication for 1 h. Then, 20 μL of the ink was dropped onto the surface of the glassy carbon electrode with a diameter of 0.5 cm. To ensure Sn-p/CN was totally converted to Sn-QDs/CN, the electrode was in-situ reduced at -1.51 V (vs. Ag/AgCl) in CO$_2$-saturated 0.1 mol·L$^{-1}$ KHCO$_3$ solution for 2 h. The as-obtained Sn-QDs/CN electrode was washed with water several times and immersed in electrolyte immediately, then was used as the working electrode for the follow-up measurements of electrochemical CO$_2$RR. The small amount of Sn-QDs/CN was scraped from the glassy carbon electrode and washed with water for further characterizations.

2 Results and discussion

As described in the experimental section, Sn-QDs/CN was synthesized by an in-situ electro-chemical-reduction process, similarly to the previously mentioned methods in the literatures$^{[17-18]}$. The crystal structure of the as-synthesized Sn-QDs/CN sample was characterized by XRD, as shown in Fig. 1. The weak peak at 2θ=27.9° corresponds to the (002) interlayer reflection of the graphitic-like structure of CN$^{[19-20]}$. The other peaks at 2θ=30.6°, 32.0°, 43.9° and 44.9° are assigned to the (200), (101), (220) and (211) planes of tetragonal Sn (JCPDS 04-0673).

Fig. 1 XRD patterns of Sn-QDs/CN and CN

Fig. 2 TEM images at different magnifications (a, b) and corresponding EDS line scanning spectra (c) of Sn-p/CN; TEM image (inset: magnified image) (d), HRTEM image (e) and Sn-QDs size distribution (f) of Sn-QDs/CN
The TEM images of Sn-p/CN in Fig. 2(a) show that Sn particles in arbitrary size (50-500 nm) are dispersed on the surface of CN. From the TEM image (Fig. 2(b)) and EDS line scanning spectra (Fig. 2(c)), a thin Sn oxide (SnOₓ) layer (light regions) forms on the surface of metallic Sn particles (dark regions), due to the oxidation of Sn in air[21]. Compared to Sn-p/CN, the Sn-QDs are homogeneously dispersed on the surface of CN substrate (Fig. 2(d)). The size distribution histogram (Fig. 2(f)) shows that Sn-QDs possess a size range of 1.6-4.4 nm. Sn species in smaller size may expose more active sites for the enhanced performance of CO₂RR[11, 22].

The highly magnified TEM image (Fig. 2(e)) reveals the good crystallinity of Sn-QDs, confirmed by the clear fringes with a lattice spacing of 0.29 nm, assigned to the (200) crystal plane of Sn (JCPDS 04-0673). On the bases of the above experimental results, Sn-QDs with an average size of only 2-3 nm was successfully decorated on CN via an in-situ electro-chemical transformation from Sn-p/CN.

The surface chemical state of the Sn-QDs/CN electrocatalyst was investigated by XPS analysis. The high-resolution Sn3d XPS spectrum (Fig. 3(a)) shows that the surface of Sn-QDs/CN consists of both Sn⁴⁺ (495.4 and 487.0 eV) and Sn⁰ (493.6 and 485.2 eV)[23]. In the O1s XPS spectrum (Fig. 3(b)), the two fitted O1s peaks at 530.6 and 533.6 eV, are assigned to O-Sn⁴⁺ and surface hydroxyl groups, respectively[24]. The existence of Sn⁴⁺ on the surface was inevitable because Sn-QDs/CN sample underwent reoxidation by exposure in air as it was transported to XPS[16]. The N1s XPS spectra (Fig. S2) can be deconvoluted into three peaks corresponding to the sp² hybridized N atoms (398.4 eV), tertiary N groups (399.7 eV) and amino groups (400.9 eV), similar with those of CN.

Compared with CN, the content of -NH₂ increases from 6.8% of CN to 15.5% of Sn-QDs/CN, indicating the slight loss of tri-s-thiazine in CN substrate. The increased amine groups could enhance the chemical adsorption of CO₂ during CO₂RR[25].

As described at experimental section, once obtained by in-situ electrochemical-reduction, the Sn-QDs/CN electrode was investigated immediately to prevent its surface reoxidation. The CO₂RR performance of the Sn-QDs/CN catalyst was explored and summarized in Fig. 4.

As shown in Fig. 4(a), a current density of about 0.6 mA·cm⁻² is obtained at -1.0 V in Ar-saturated solution, which is ascribed to HER (Hydrogen evolution reaction). By comparison, in CO₂-saturated solution, the current density is ~2.0 mA·cm⁻² at -1.0 V, which indicates that Sn-QDs/CN exhibits much higher CO₂RR activity than HER[26]. Fig. S3(a) shows that the current density of Sn-QDs/CN during CO₂RR increases with the rise of applied potentials, coinciding with the result of LSV in Fig. 4(a).

The Faradaic efficiencies of products at different applied potentials are plotted to evaluate the selectivity of the Sn-QDs/CN electrode (Fig. S3(b)). CO and HCOOH are generated in CO₂ electroreduction reaction. H₂ is produced from HER, the competing reaction with CO₂RR. On CN substrate (Fig. S4), no HCOOH is generated during CO₂RR, and H₂ is the major product. This signifies that the active sites of CO₂RR on Sn-QDs/CN are Sn-QDs. As revealed by Fig. S3(b), the Faradaic efficiency of HCOOH (FEHCOOH) exceeds 83% in a record wide potential window.
of 0.5 V (from -0.8 V to -1.3 V), manifesting its high selectivity for HCOOH. To be specific, the Sn-QDs/CN electrode displays the maximum FE$_{\text{HCOOH}}$ of around 95% at -1.0 V. While, Sn-p/CN exhibits a maximum FE$_{\text{HCOOH}}$ of only 75% at -1.0 V. Faradaic efficiencies of HCOOH on Sn-QDs/CN and Sn-p/CN are intuitively compared in Fig. 4(b) at a series of potentials. It can be observed clearly that Sn-QDs/CN shows much more outstanding CO$_2$RR performance for HCOOH production at all applied potentials. Therefore, it can be concluded that the performance of CO$_2$RR has a great relationship with the particle size. Moreover, the as-prepared Sn-QDs/CN electrode in CO$_2$-saturated 0.1 mol·L$^{-1}$ KHCO$_3$ solution exhibits enhanced current density compared to Sn-p/CN in the examined potential range (Fig. S5), indicating that the catalytic activity on CO$_2$RR of Sn-QDs/CN is much higher than that on Sn-p/CN.

To explore the effect of particle size on the performance of CO$_2$ reduction, the double-layer capacitance was measured to estimate the ECSA of catalyst. As shown in Fig. 5(a), Sn-QDs/CN presents higher capacitance (0.053 mF·cm$^{-2}$) than Sn-p/CN (0.012 mF·cm$^{-2}$), which indicates that the ECSA of Sn-QDs/CN is about 4.4 times of that of Sn-p/CN. The increased ECSA means more active sites exposed for CO$_2$RR, contributing to the enhanced catalytic performance. Moreover, Sn-QDs/CN possesses a smaller radius of impedance than Sn-p/CN (Fig. 5(b)). It can be seen from Table S2 that the fitted charge transfer resistance ($R_{\text{ct}}$) of Sn-QDs/CN is 276.3 Ω, which is much smaller than that of Sn-p/CN (336.9 Ω). It manifests that the charge-transfer impedance in Sn-QDs/CN is much smaller than that in Sn-p/CN, i.e., much faster electron transfer in Sn-QDs/CN catalyst during CO$_2$RR.

To better explore the kinetic mechanism of CO$_2$RR, Tafel slope was further analysed. Tafel plots for HCOOH production are presented in Fig. 5(c). The measured Tafel slope values of Sn-QDs/CN and Sn-p/CN are 170 and 250 mV·dec$^{-1}$, respectively, which are both close to the theoretical value of 116 mV·dec$^{-1}$. The data imply that the rate-determining step (RDS) on these two catalysts for CO$_2$RR is supposed to be one electron transfer to CO$_2$ to form CO$_2^-$ intermediate$^{[26]}$. Besides, the lower Tafel slope of Sn-QDs/CN indicates its much faster CO$_2$RR kinetics than Sn-p/CN, consistent with the result of EIS.

Except for the high electrocatalytic performance for HCOOH production in CO$_2$RR, Sn-QDs/CN electrocatalyst also showed good electrochemical durability. In Fig. 5(d), it can be found that both the FE$_{\text{HCOOH}}$ and the current density display no obvious decay during the continuous CO$_2$RR at -1.0 V for 24 h. The electrochemical stability implies the favorably stable structure of Sn-QDs/CN.

![Fig. 5](image)

**Fig. 5** Charging current density differences plotted against scan rates (a), electrochemical impedance spectra with inset showing the corresponding equivalent circuit (b), Tafel plots for HCOOH production on Sn-QDs/CN and Sn-p/CN (c), and the stability of Sn-QDs/CN catalyst at -1.0 V for 24 h in CO$_2$-saturated 0.1 mol·L$^{-1}$ KHCO$_3$ (d)

## 3 Conclusion

In summary, we have successfully fabricated an electrocatalyst (Sn-QDs/CN) with Sn quantum dots (Sn-QDs) decorated on polymeric carbon nitride (CN) by an *in-situ* electrochemical reduction process conducted on Sn particles loaded CN (Sn-p/CN) in a three-electrode system. The Sn-QDs with good crystallinity exhibit an average size of only 2-3 nm. The smaller size brings more active sites exposed, and the electrochemical active surface area (ECSA) of Sn-QDs/CN also increases by about 3.4 times in comparison with Sn-p/CN. The Faradic efficiency of HCOOH (FE$_{\text{HCOOH}}$) reached up to 95% on Sn-QDs/CN at -1.0 V and kept above 83% in a recorded wide potential window exceeding 0.5 V. Furthermore, the Sn-QDs/CN electrocatalyst also showed good electrochemical stability during the CO$_2$RR of 24 h. The advanced performance of Sn-QDs/CN can be attributed to the enlarged ECSA, fastened electron transfer and enhanced kinetics derived from the decreased particle size of Sn-QDs. This work provides...
a perspective to synthesize metal QDs electrocatalysts for highly efficient CO$_2$RR to HCOOH.

**Supporting materials**

Supporting materials related to this article can be found at https://doi.org/10.15541/jim20210177

**Reference:**

[1] LI X D, WANG S M, LI L, et al. Progress and perspective for in situ studies of CO$_2$ reduction. *Journal of the American Chemical Society*, 2020, 142: 9567-9581.

[2] BIRDIA Y Y, PEREZ-GALLENT E, FIGUEIREDO M, et al. Advances and challenges in understanding the electrocatalytic conversion of carbon dioxide to fuels. *Nature Energy*, 2019, 4: 732-745.

[3] VASILEFF A, ZHENG Y, QIAO S Z. Carbon solving carbon's problems: recent progress of nanostructured carbon-based catalysts for the electrochemical reduction of CO$_2$. *Advanced Energy Materials*, 2017, 7(21): 1700759.

[4] SHAO P, YI L C, CHEN S M, et al. Metal-organic frameworks for electrochemical reduction of carbon dioxide: the role of metal centers. *Journal of Energy Chemistry*, 2020, 40: 156-170.

[5] ZHANG L, ZHAO Z J, GONG J. Nanostructured materials for heterogeneous electrocatalytic CO$_2$ reduction and their related reaction mechanisms. *Angewandte Chemie International Edition*, 2017, 56(38): 11326-11353.

[6] MISTRY H, RESKE R, ZENG Z H, et al. Exceptional size-dependent activity enhancement in the electrodreduction of CO$_2$ over Au nanoparticles. *Journal of the American Chemical Society*, 2014, 136(47): 16473-16476.

[7] TYO E C, VAIDYA S. Catalysis by clusters with precise numbers of atoms. *Nature Nanotechnology*, 2015, 10(7): 577-588.

[8] GAO D F, ZHOU H, WANG J, et al. Size-dependent electrocatalytic reduction of CO$_2$ over Pd nanoparticles. *Journal of the American Chemical Society*, 2015, 137(13): 4288-4291.

[9] LIU S G, HUANG S P. Size effects and active sites of Cu nanoparticle catalysts for CO$_2$ electroreduction. *Applied Surface Science*, 2019, 475: 20-27.

[10] LEE C H, KANAN M W. Controlling H$_2$ vs CO$_2$ reduction selectivity on Pb electrodes. *ACS Catalysis*, 2014, 5(1): 465-469.

[11] LI Z D, HE D, YAN X X, et al. Size-dependent nickel-based electrocatalysts for selective CO$_2$ reduction. *Angewandte Chemie International Edition*, 2020, 59: 2-8.

[12] RESKE R, MISTRY H, BEHAFARID F, et al. Particle size effects in the catalytic electrodreduction of CO$_2$ on Cu nanoparticles. *Journal of the American Chemical Society*, 2014, 136(19): 6978-6986.

[13] LÜ K L, SUO W Q, SHAO M D, et al. Nitrogen doped MoS$_2$ and nitrogen doped carbon dots composite catalyst for electro-reduction CO$_2$ to CO with high Faradaic efficiency. *Nano Energy*, 2019, 63: 103834.

[14] LIU M, LIU M X, WANG X M, et al. Quantum-dot-derived catalysts for CO$_2$ reduction reaction. *Joule*, 2019, 3(7): 1703-1718.

[15] WU J J, MA S C, SUN J, et al. A metal-free electrocatalyst for carbon dioxide reduction to multi-carbon hydrocarbons and oxygenates. *Nature Communication*, 2016, 7: 13869.

[16] TIAN J J, WANG M, SHEN M, et al. Highly efficient and selective CO$_2$ electro-reduction to HCOOH on Sn particle-decorated polymeric carbon nitride. *ChemSusChem*, 2020, 13(23): 6442-6448.

[17] WEN G B, LEE D U, REN B H, et al. Orbital interactions in Bi-Sn bimetallic electrocatalysts for highly selective electrochemical CO$_2$ reduction toward formate production. *Advanced Energy Materials*, 2018, 8(31): 1802427.

[18] LI P X, FU W Z, ZHUANG P Y, et al. Amorphous Sn/crystalline Sn$_2$ nanosheets via in situ electrochemical reduction methodology for highly efficient ambient N$_2$ fixation. *Small*, 2019, 15(40): 1902535.

[19] TIAN J J, ZHANG L X, WANG M, et al. Remarkably enhanced H$_2$ evolution activity of oxidized graphitic carbon nitride by an extremely facile K$_x$CO$_3$-activation approach. *Applied Catalysis B: Environmental*, 2018, 232: 322-329.

[20] WEN J, XIE J, CHEN X, et al. A review on g-C$_3$N$_4$-based photocatalysts. *Applied Surface Science*, 2017, 391: 72-123.

[21] LAI Q, YUAN W Y, HUANG W J, et al. Sn/SnO$_2$ electrode catalyst with mesoporous structure for efficient electrodreduction of CO$_2$ to formate. *Applied Surface Science*, 2020, 508: 145221.

[22] LIU S B, XIAO J, LU X F, et al. Efficient electrochemical reduction of CO$_2$ to HCOOH over Sub-2 nm SnO$_2$ quantum wires with exposed grain boundaries. *Angewandte Chemie International Edition*, 2019, 58: 8499-8503.

[23] LUC W, COLLINS C, WANG S W, et al. Ag-Sn bimetallic catalyst with a core-shell structure for CO$_2$ reduction. *Journal of the American Chemical Society*, 2017, 139(5): 1885-1893.

[24] BANG J H, CHOI M S, MIRZAEI A, et al. Selective NO$_3$ sensor based on Bi$_2$O$_3$ branched SnO$_2$ nanowires. *Sensors and Actuators B: Chemical*, 2018, 274: 356-369.

[25] MA Y, WANG Z, XU X, et al. Review on porous nanomaterials for adsorption and photocatalytic conversion of CO$_2$. *Chinese Journal of Catalysis*, 2017, 38(12): 1956-1969.

[26] CHEN Z, GAO M R, DUAN N, et al. Tuning adsorption strength of CO$_2$ and its intermediates on tin oxide-based electrocatalyst for efficient CO$_2$ reduction towards carbonaceous products. *Applied Catalysis B: Environmental*, 2020, 277: 119252.

[27] NGUYEN T N, SALEHI M, LE Q, et al. Fundamentals of electrochemical CO$_2$ reduction on single-metal-atom catalysts. *ACS
锡量子点制备及其电催化还原二氧化碳产甲酸性能

摘 要：锡基材料在自然界含量丰富、价格低廉，在电催化还原 CO₂ 制液体燃料反应中具有巨大潜力。但是较低的产物选择性和较差的稳定性限制了其应用。本工作制备了锡量子点催化剂 (Sn-QDs)，具有高效、高稳定性和高选择性的电催化还原 CO₂ 产 HCOOH 活性。Sn-QDs 的平均颗粒尺寸仅为 2~3 nm，结晶性良好。小的颗粒尺寸增大了电化学活性面积 (ECSA)，Sn-QDs 的 ECSA 约为锡颗粒的 4.4 倍。ECSA 增大以及 CO₂ 还原反应动力学加速，促进了 CO₂ 电化学转化。在 -1.0 V (vs RHE) 下，Sn-QDs/CN 催化剂的 HCOOH 法拉第效率 (FE_HCOOH) 达到 95%。并且在宽约 0.5 V 的电势范围内能够保持在 83% 以上。此外，Sn-QDs/CN 可以在 24 h 内保持良好的电化学稳定性。

关键词：CO₂ 还原；锡量子点；HCOOH；电催化剂

中图分类号：TQ174 文献标志码：A
Supporting materials

Sn Quantum Dots for Electrocatalytic Reduction of CO₂ to HCOOH

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1 Characterizations
X-ray diffraction (XRD) patterns were acquired on a Rigaku D/Max 2200PC diffractometer (Cu Kα radiation). The X-ray photoelectron spectroscopy (XPS) measurement was conducted on a Thermo Scientific ESCAlab 250 spectrometers with Al Kα X-ray as radiation source, and the binding energies were calibrated using the C1s peak at 284.6 eV. The transmission electron microscope (TEM) and high resolution TEM (HRTEM) images were obtained on a JEM-2100F microscope.

2 Electrochemical measurements
All electrochemical measurements were performed in a gas-tight H-cell (separated by Nafion N117 membrane) connected to a Biologic VMP3 electrochemistry workstation. Each of cathodic and anodic compartments held 30 mL of electrolyte (CO₂-saturated 0.1 mol·L⁻¹ KHCO₃) and 38 mL of headspace. The Pt mesh and Ag/AgCl electrode (with a saturated KCl solution) were used as the counter electrode and reference electrode, respectively. All measured potentials were converted to the reversible hydrogen electrode (RHE) without special illustration in the whole work. The linear sweep voltammetry (LSV) was recorded at a scan rate of 30 mV·s⁻¹. The current density was calculated by normalizing to the geometric surface area of the working electrode. Electrochemical impedance spectroscopy (EIS) measurement was carried out in the frequency range from 100 KHz to 10 Hz at -1.0 V (vs RHE). Double-layer capacitance (Cdl) was determined by measuring the capacitive current associated with double-layer charging from the scan-rate dependence of cyclic voltammetry (CV). The potential window of CV was 0.0 V to -0.1 V (vs RHE). The scan rates were 10-100 mV·s⁻¹. The Cdl was estimated by plotting the Δj (Δj = jₐ-jₖ) at -0.05 V (vs RHE) against the scan rates (jₐ and jₖ were the anodic and cathodic current densities, respectively). The slope was twice of that of Cdl.

Electrochemical CO₂ reduction was conducted at room temperature and atmospheric pressure. Prior to CO₂RR, high purity CO₂ gas (99.99%) was firstly purged into the cathodic compartment reservoir for 30 min to remove residual air, then the cathodic compartment was sealed. The anode chamber was open in order to vent out O₂ generated during the electrochemical test. The electrolyte in the cathode chamber was continuously stirred at a rate of 300 r·min⁻¹ during the electrolysis to enhance the mass transport of CO₂. The CO₂RR was conducted for 2 h at each potential. The concentration of gaseous products was detected by gas chromatography (GC7900 Techcomp for H₂ and GC-2014 Shimadzu for CO). Liquid products were analyzed on a 500 MHz ¹H liquid NMR spectrometer (Bruker) with dimethyl sulfoxide (DMSO) as internal standard (Fig. S1(a)). The ¹H NMR spectrum was recorded using water suppression method. After electrolysis, the product-containing electrolyte (500 μL) was mixed with 100 μL of D₂O solution consisting of 12.46 mmol·L⁻¹ DMSO for NMR test. The concentration of formate (HCOOH) was obtained based on the calibration curve (Fig. S1(b)). The Faradaic efficiency of the product was calculated as below: \( \text{FE}_{\text{product}} = \frac{2F \times n_{\text{product}}}{Q} \), where \( F \) is the Faraday constant (96485 C·mol⁻¹), \( n \) is the amount of production (mol), \( Q \) is the total charge consumed (C).
Table S1 Comparison of various Sn-based catalysts for CO$_2$-to-HCOOH conversion

| Electro catalyst                  | Electrolyte                | Potential (/V vs. RHE) | FE$_{\text{HCOOH}}$ (%) | Current density (mA·cm$^{-2}$) | Stability /h | Ref.         |
|----------------------------------|---------------------------|------------------------|--------------------------|-------------------------------|--------------|--------------|
| Sn-QDs/CN                        | 0.1 mol·L$^{-1}$ KHCO$_3$  | -1.0                   | 95                       | 3.3                           | 24           | This work    |
| Sn quantum sheets confined in graphene | 0.1 mol·L$^{-1}$ NaHCO$_3$ | -1.2                   | 89                       | 21.1                          | 50           | [1]          |
| nano-SnO$_2$/graphene            | 0.1 mol·L$^{-1}$ NaHCO$_3$ | -1.2                   | 93.6                     | 10                            | -            | [2]          |
| SnO$_2$ nanoparticles (< 5 nm)    | 0.1 mol·L$^{-1}$ KHCO$_3$  | -1.2                   | 64                       | 147                           | -            | [3]          |
| SnO$_2$ nanoparticles (~500 nm)   | 0.1 mol·L$^{-1}$ KHCO$_3$  | -1.2                   | 83.5                     | 7.56                          | -            | [4]          |
| SnO$_2$ nanoparticles (100 nm)    | 0.5 mol·L$^{-1}$ KHCO$_3$  | -0.9                   | 80                       | 12                            | -            | [5]          |
| SnO$_2$ nanoparticles (8-20 nm)   | 0.1 mol·L$^{-1}$ KHCO$_3$  | -1.06                  | 82                       | 15.3                          | 5            | [6]          |
| SnO$_2$@N-CNW                    | 0.5 mol·L$^{-1}$ NaHCO$_3$ | -0.8                   | 90                       | 13                            | 20           | [7]          |
| SnO$_2$@N-rGO                    | 0.5 mol·L$^{-1}$ NaHCO$_3$ | -0.8                   | 89                       | 21.3                          | 20           | [8]          |
| SnO$_2$/PC                       | 0.5 mol·L$^{-1}$ KHCO$_3$  | -0.86                  | 92                       | 29                            | 10           | [9]          |
| SnO$_2$⊃NC@EEG                   | 0.1 mol·L$^{-1}$ KHCO$_3$  | -1.2                   | 81.2                     | 13.4                          | 10           | [10]         |
| SnO/C                            | 0.5 mol·L$^{-1}$ KHCO$_3$  | -0.86                  | 75                       | 27.2                          | -            | [11]         |

**Fig. S1** $^1$H NMR spectrum of the cathodic electrolyte after CO$_2$RR (a), and linear relationship between HCOOH concentration and relative peak area ratio (vs. DMSO) (b)
**Fig. S2** High-resolution N1s XPS spectra of Sn-QDs/CN

**Fig. S3** $i$-$t$ curves of CO$_2$RR on Sn-QDs/CN at different applied potentials (a), and Faradaic efficiencies of HCOOH, CO and H$_2$ at different applied potentials on the Sn-QDs/CN electrode (b)

**Table S2** The fitted data of EIS for Sn-QDs/CN and Sn-p/CN

| Parameter | $R_s/\Omega$ | $R_o/\Omega$ | CPE-T | CPE-P |
|-----------|--------------|--------------|--------|--------|
| Sn-QDs/CN | 102.6        | 276.3        | $1.2 \times 10^{-5}$ | 0.85   |
| Sn-p/CN   | 96.74        | 336.9        | $1.1 \times 10^{-5}$ | 0.89   |

**Fig. S4** Faradaic efficiencies of CO and H$_2$ at different applied potentials on CN
Fig. S5 LSV curves of Sn-p/CN and Sn-QDs/CN in CO₂-saturated 0.1 mol·L⁻¹ KHCO₃ electrolyte at a scan rate of 30 mV·s⁻¹

\[ \text{CO}_2 + e^- \rightarrow \text{CO}_2^- \]

\[ \text{CO}_2^- + H^+ + e^- \rightarrow \text{HCOO}^- \]

\[ \text{HCOO}^- + H^+ + e^- \rightarrow \text{HCOOH} \]

Fig S6. Proposed possible reaction pathway of CO₂-to-HCOOH conversion on Sn-QDs/CN

References:
[1] LEI F C, SUN Y F, XU J Q, et al. Metallic tin quantum sheets confined in graphene toward high-efficiency carbon dioxide electroreduction. *Nature Communications*, 2016, 7: 12697.
[2] ZHANG S, KANG P, J. MEYER T. Nanostructured tin catalysts for selective electrochemical reduction of carbon dioxide to formate, *Journal of the American Chemical Society*, 2014, 136: 1734-1737.
[3] LIANG C L, KIM B, YANG S Z, et al. High efficiency electrochemical reduction of CO₂ beyond the two-electron transfer pathway on grain boundary rich ultra-small SnO₂ nanoparticles. *Journal of Materials Chemistry A*, 2018, 6: 10313-10319.
[4] WANG Y, ZHOU J, LV W X, et al. Electrochemical reduction of CO₂ to formate catalyzed by electroplated tin coating on copper foam. *Applied Surface Science*, 2016, 362: 394-398.
[5] WU J J, P. SHARMA P, HARRIS B H, et al. Electrochemical reduction of carbon dioxide: IV dependence of the Faradaic efficiency and current density on the microstructure and thickness of tin electrode. *Journal of Power Sources*, 2014, 258: 189-194.
[6] BEJTKA K, ZENG J Q, SACCO A, et al. Chainlike mesoporous SnO₂ as a well-performing catalyst for electrochemical CO₂ reduction. *ACS Applied Energy Materials*, 2019, 2: 3081-3091.
[7] ZHANG B H, SUN L Z, WANG Y Q, et al. Well-dispersed SnO₂ nanocrystals on N-doped carbon nanowires as efficient electrocatalysts for carbon dioxide reduction. *Journal of Energy Chemistry*, 2020, 41: 7-14.
[8] ZHANG B H, GUO Z H, ZUO Z, et al. The ensemble effect of nitrogen doping and ultrasmall SnO₂ nanocrystals on graphene sheets for efficient electroreduction of carbon dioxide. *Applied Catalysis B: Environmental*, 2018, 239: 441-449.
[9] HE Y H, JIANG W J, ZHANG W J, et al. Pore-structure-directed CO₂ electroreduction to formate on SnO₂/C catalysts. *Journal of Materials Chemistry A*, 2019, 7: 18428-18433.
[10] FU Y Y, WANG T T, ZHENG W Z, et al. Nanoconfined tin oxide within N-doped nanocarbon supported on electrochemically exfoliated graphene for efficient electroreduction of CO₂ to formate and C₁ products. *ACS Applied Materials & Interfaces*, 2020, 12: 16178-16185.
[11] GU J, HEROGUEL F, LUTERBACHER J, et al. Densely packed, ultra small SnO nanoparticles for enhanced activity and selectivity in electrochemical CO₂ reduction. *Angewandte Chemie International Edition*, 2018, 57: 2943-2947.