State-of-the-art of chemically grown vanadium pentoxide nanostructures with enhanced electrochemical properties

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ABSTRACT

This review begins by the principles that describe the intercalation properties of vanadium pentoxide. Recent developments in the chemical synthesis of vanadium pentoxide nanostructures including nanorod and nanotube arrays, nanowires, anhydrous and hydrous nanorolls and nanobelts, nanoribbons, nanocables and gels are examined. The objective of this review is to demonstrate the use of these nanostructures as cathodes for lithium batteries and electrochromics with high storage capacity and rate performance due to the large surface area and short distance for charge transport compared to the thin film cathodes. Copyright © 2013 VBRI press.

Keywords: Vanadium pentoxide; nanostructures; intercalation properties; batteries; electrochromics.

1. Introduction

Vanadium pentoxide is a typical intercalation compound because of the multiple valence state of vanadium and the rich structural chemistry, which enables redox-dependent properties [1]. The intercalation refers to the reversible insertion of guest species into a host lattice that contains an interconnected system of an empty lattice site of appropriate size, while the structural integrity of the host lattice is formally conserved. When guest species are incorporated into host lattices, various structural changes take place. The principle geometrical transitions of host lattice matrixes upon intercalation of guest species include: a) change in interlayer spacing, b) change in stacking mode of the layers and c) formation of intermediate phases at low guest concentrations that may exhibit staging [2].

The intercalation reactions take place with a large free energy of formation, which has led to their use as the cathodes in lithium batteries [3]. Concomitant with the species insertion, electrons are introduced into the conduction bands giving rise to changes in colour. This ready colouration under ambient conditions led to their use in electrochromic displays [4].

This review will be concerned with the recent developments in the synthesis of nanostructured vanadium pentoxide. The relationship between the structure and properties with the electrochemical behaviour as relevant to batteries and electrochromics will be discussed. The impact of morphology and size will be described as there is much interest in nano-sized structures for diffusion controlled...
1.1. How do the lithium batteries operate?

A battery is a device that converts chemical potential to electric energy through Faradaic reactions, which includes heterogeneous charge transfer occurring at the surface of an electrode [5]. It consists of three basic components: an anode, a cathode and an electrolyte. It is broadly grouped as primary and secondary battery. A primary battery is a single-use device and cannot be recharged, while a secondary battery can be recharged for many times.

The principal concept of lithium batteries is shown in Fig. 1. An anode (source of the lithium ions) with a cathode (sink for the lithium ions) having a more positive redox potential gives a lithium transfer cell. They are separated by the electrolyte, which is a lithium ion conductor. It provides the separation of ionic and electronic transport. Upon charging, lithium ions are released by the cathode and intercalated at the anode. While upon discharging, lithium ions are extracted by the anode and inserted into the cathode. Cathode materials must fulfill the following requirements to reach the goal of a high energy density (expressed in Wh L$^{-1}$) [6]: (1) a high number of available charge carriers per mass and volume unit, (2) a high cell voltage, resulting from a high (cathode) and low (anode) standard redox potential of the respective cathode redox reaction, (3) a high reversibility of electrochemical reactions at both cathodes and anodes to maintain the specific charge for hundreds of charge-discharge cycles, (4) stability, i.e. no degradation upon cycling, (5) low cost and (5) environmentally benign.

![Fig. 1. A schematic presentation of a lithium battery, taken from [7].](image)

Currently there are three intercalation materials that are used commercially as cathode materials in lithium batteries: LiCoO$_2$, LiNiO$_2$ and LiMn$_2$O$_4$. Among those, LiCoO$_2$ is the most popular because of the simplicity of its preparation [8]. The Li$_x$CoO$_2$ system has been studied extensively thus far exhibiting excellent cyclability at room temperature for 1$\times$10$^{5}$. The specific capacity of the material is limited to the range 137–140 mAh g$^{-1}$, although the theoretical capacity of LiCoO$_2$ is 273 mAh g$^{-1}$ [9]. Even though it has good electrochemical properties and is easy to synthesize, it is very expensive and highly toxic. On the other hand, the reversible capacity of LiNiO$_2$ is higher than that of Li$_x$CoO$_2$, since the amount of lithium that can be intercalated / extracted during redox cycles is around 0.55 in comparison with 0.5 for LiCoO$_2$, allowing the specific capacity to be more than 150 mAh g$^{-1}$ with appropriate cyclability [10]. Although LiNiO$_2$ and LiCoO$_2$ are isostructural, the preparation of LiNiO$_2$ is more complicated. Since there are additional nickel ions on the lithium sites and vice versa in the crystal structure of LiNiO$_2$, the Li–Ni–O system is represented by Li$_{1-x}$Ni$_{x}$O$_2$, a deviation from the normal stoichiometry [11]. In comparison with LiCoO$_2$ and LiNiO$_2$, LiMn$_2$O$_4$ is less toxic and has an abundant source. In principle, Li$_x$Mn$_2$O$_4$ permits the intercalation / extraction of lithium ions in the range 0<x<2 [12]. For values of x between 1 and 2, the material consists of two different phases—cubic in the bulk and tetragonal at the surface. Simultaneously, the intercalation of lithium ions effectively decreases the average valence of manganese ions and leads to a pronounced cooperative Jahn–Teller effect, in which the cubic spinel crystal becomes distorted tetragonal with c/a $\approx$ 1.16, and the volume of the unit cell increases by 6.5 %. This high c/a ratio causes a low capacity restricted to $<120–125$ mAh g$^{-1}$ and significant capacity degradation at moderate temperatures in the range of 50–70 °C [13].

The limitations of the cathode materials have stimulated research activity targeted at investigating new materials for lithium batteries [14]. Vanadium pentoxide has attracted much interest because it offers the advantages of being cheap, easy to synthesize and having high energy densities. Electrochemical lithium ion intercalation occurs together with compensating electron intercalation, leading to the formation of vanadium bronzes as follows in equation (1):

$$V_2O_5+xl^+ + xe^- \leftrightarrow Li_xV_2O_5$$ (1)

However, the intercalation capacity and charge / discharge rate of V$_2$O$_5$ are limited by the low diffusion coefficient of Li ions ($10^{-12}$–$10^{-13}$ cm$^2$ s$^{-1}$) in the V$_2$O$_5$ matrix [15]. Many studies have been conducted to improve lithium diffusion and electrical conduction performance in V$_2$O$_5$ including nanostructured materials that possess large surface areas and short diffusion paths. Ordered arrays of nanorods, nanotubes or nanocables are some of the most promising nanostructures for lithium intercalation applications. Patrissi and Martin investigated the electrochemical properties of V$_2$O$_5$ nanorod arrays, which achieved four times the capacity of a thin-film cathode at high discharge rate [16]. Compared to nanorods, nanotubes possess higher surface area due to their several different areas of contact, i.e., the inner and outer wall surfaces, as well as the open ends. In addition, they can operate as electrolyte-filled channels for faster transport of the ions to the intercalation sites.

Another approach to increase the intercalation capacity of vanadium pentoxide is to modify its interlayer structure and the interaction forces between the adjacent layers. When the distance between the adjacent V$_2$O$_5$ layers increases, the insertion capacity increases. For example, V$_2$O$_5$·nH$_2$O possesses a lithium intercalation capacity that is about 1.4 times larger than that of V$_2$O$_5$ [17]. The distance between the adjacent layers in V$_2$O$_5$·nH$_2$O is 11.52 Å [18], compared to an interlayer distance of 4.56 Å in orthorhombic V$_2$O$_5$. Doping or substitution in V$_2$O$_5$ by other cations with different valence states have been used to tailor the interaction forces between two adjacent layers in the intercalation compound. Several materials, such as Ni...
1.2. What are the electrochromics?

Electrochromics can be darkened (reflecting light) or lightened (transmitting light) in response to an electric field (Fig. 2). They are made by sandwiching certain materials between two panes of glass or plastic as listed below:

- Glass or plastic panel
- Conducting oxide
- Electrochromic layer
- Ion conductor / electrolyte
- Ion storage
- Conducting oxide
- Glass or plastic panel

![Fig. 2. (a) When the system is switched off, the electrochromic window is transparent. (b) When the system is switched on, the electrochromic window is dark, taken from [22].](image)

The general electrochromic requirements are the following:

- A minimum contrast ratio 5:1 of the clear state (bleached) to the darkened (colored) state.
- Good cyclic lifetime (>100000 cycles).
- Continuous range in heat and optical transmittance, reflectance and absorbance between bleached and coloured states.
- Colouring and bleaching times of a few seconds.
- Switching with applied voltages 1-5 V.
- Large area with optical clarity.
- Sustained performance over 20-30 years.
- Acceptable cost (70 €/m²).

Numerous transition metal oxides possess the electrochromic property [4]: a decrease of transmittance occurs upon ion insertion in oxides of Ti, Nb, Mo, Ta and W, while an increase of the transmittance upon ion insertion in oxides of Cr, Mn, Fe, Co, Ni, Rh and Ir. Among those, tungsten trioxide appears to be the best electrochromic material. Its advantages are high colouration efficiency, reasonable stability and relatively low cost. Nevertheless, for building applications, the bright blue colour of WO₃ in the reduced state is not as favourable as neutral gray or bronze colours. Hence, the need for materials exhibiting a different colour from that of WO₃ is important to extend the range of functions of electrochromics.

In this case, electrochromic vanadium pentoxide has attracted a lot of interest. This oxide is special in that charge insertion increases the transmittance in the ultraviolet and short-wavelength parts of the luminous spectrum, while it drops the transmittance in the long-wavelength part of this spectrum as well as in the near infrared [23-31].

However, the kinetics of ion insertion is limited by solid state lithium ion diffusion. Reducing the length of the diffusion path by designing a highly porous network is a desirable approach to address this issue. In the case of vanadium pentoxide, gels serve as host materials for a wide variety of metal cations [32]. The large lithium insertion capacity of vanadium pentoxide xerogels, theoretically up to two equivalent of lithium per vanadium, makes them attractive for use as cathodes in electrochromic devices. In addition, the capacity of lithium intercalation at high discharge rates can be improved by controlling the size and shape of the individual particles and the morphology of the V₂O₅. For example, the rate capabilities of the nanostructured cathodes were compared with the rate capabilities of thin film cathodes showing that the nanostructured cathodes delivered higher capacities than the thin film cathodes [2]. This is because the nanostructured cathode has a short Li-ion insertion / extraction distance, facile strain relaxation upon electrochemical cycling, and very large surface to volume ratio to contact with the electrolyte, which can improve its capacity and cycle life. Hierarchical nanostructured materials such as hollow nanospheres, porous nanostructures, nanotubes, nanowire structures, and kinked nanowires, etc., can ensure that surface remains uncovered to keep the effective contact areas large even if a small amount of inevitable self-aggregation occurs. Moreover, if one dimension of the nanocrystallites is up to a few hundred micrometers or even at millimeter scale, such as ultralong nanowires or nanobelts, self-aggregation of the nanomaterials can be effectively prevented. Therefore, to some extent, an ultralong hierarchical nanowire is one of the most favourable structures as cathode materials.

2. Characteristics of vanadium pentoxide

2.1. Structure and properties of crystalline V₂O₅ upon lithium insertion

Orthorhombic crystalline V₂O₅ consists of layers of VO₅ square pyramids that share edges and corners [12] (see Fig. 3). The vanadium-oxygen bond distance is much shorter than the four other distances and corresponds to a double bond illustrating a two-dimensional character of this material. In addition, the structure of V₂O₅ can be described as distorted VO₅ octahedral, which emphasizes the structural relationship to ReO₃ via crystallographic shears [33]. The very large length of the sixth vanadium-oxygen bond underlines the structural anisotropy of this material and the ability to insert guest species perovskite-like cavities.
In addition, \(\varepsilon\)-phase forms for \(0.35 < x < 0.7\) in \(\text{Li}_x\text{V}_2\text{O}_5\). Both phases exhibit a \(\text{V}_2\text{O}_5\)-type structure with increasing puckering of the layers [36]. Maximum puckering occurs when the puckering angle of the basal oxygen plane (\(\mu\)) is raised at an angle of 60°. Ideally, \(\mu\) varies from 0° up to 30° for \(\text{V}_2\text{O}_5\) (see Fig. 5).

![Fig. 3. Schematic presentation of \(\text{V}_2\text{O}_5\) layered structure. The grey circle presents vanadium centres, yellow–one coordinate oxygen atoms, red–two coordinate oxygen atoms and purple–three coordinate oxygen atoms, taken from [34].](image)

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For the composition \(\text{LiV}_2\text{O}_5\) gliding of one layer out of two, leads to the \(\delta\)-phase (see Fig. 6). This layered structure has been determined by Cava et al. [37]. It consists of [\(\text{V}_2\text{O}_5\)]\(_n\) layers of the \(\text{V}_2\text{O}_5\) type, alternatively shifted by \(b/2\) along [010]. This implies a doubling of the \(c\) parameter.

![Fig. 4. Presentation of the \(\alpha\)-phase. Triangles present \(\text{VO}_3\) square pyramids and circles \(M (= \text{e.g. lithium})\) atoms, taken from [35].](image)

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Fig. 6. Presentation of the \(\delta\)-phase. Lithium atoms are tetrahedrally surrounded, taken from [35].

![Fig. 5. Puckering of \(\text{V}_2\text{O}_5\) layers, taken from [35].](image)

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![Fig. 7. Presentation of the \(\gamma\)-phase. The lithium between layers is octahedrally surrounded by oxygen, taken from [35].](image)

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All the phase transitions are fully reversible and the pristine \(\text{V}_2\text{O}_5\) phase is recovered upon lithium extraction. However, for lithium content \(x > 1\), the \(\delta\)-phase is irreversibly transformed via a reconstruction mechanism, leading to a structural modification related to the \(\gamma\)-phase [36] (see Fig. 7). It was shown in 1989 that the resulting \(\gamma\)-phase can be reversibly cycled in the stoichiometric range \(0 < x < 2\) under retention of the \(\gamma\)-type structure [38]. This structure exhibits puckered [\(\text{V}_2\text{O}_5\)]\(_n\) layers in which the electronic localization has been established. Thus, two [\(\text{V}^{5+}\text{O}_3\)] and [\(\text{V}^{4+}\text{O}_3\)] square pyramids of widely different sizes were observed [35].

Further insertion of a third lithium into \(\text{V}_2\text{O}_5\) leads to the irreversible formation of the so-called \(\omega\)-phase with a rock-salt type structure. The resulted compound
demonstrated a specific energy up to 900 Wh kg\(^{-1}\) and 100 cycles with more than 450 Wh kg\(^{-1}\) cell in a voltage range between 3.4 and 1.9 V. Thus, it showed great potential as a cathode material for lithium batteries, which has been demonstrated for a \(\omega\)-Li\(_2\)V\(_2\)O\(_5\)/Li.

### 2.2. \(V_2O_5\cdot nH_2O\) gels

Besides \(V_2O_5\), promising results have also been reported for \(V_2O_5\cdot nH_2O\) xerogels (see Fig. 8). These structures offer considerable advantages due to their morphology. A large electrochemically active surface area, small particle size, and low density provide both high overall diffusion coefficients and low volume expansion during lithium intercalation [2]. Specific energies of over 700 Wh kg\(^{-1}\) were measured for lithium cells with a xerogel cathode [39]. However, limited long-term cycling stability is a major problem of such cathode materials at present.

![Structure of \(V_2O_5\cdot nH_2O\) gels, taken from [40].](image)

Despite decades of experimentation with \(V_2O_5\cdot nH_2O\), its atomic structure has remained a mystery because it does not form crystals and exists only as ribbonlike particles about 10 nm wide and 1 \(\mu\)m long [41]. Livage et al. proposed that the xerogel, on the atomic scale, is a stack of corrugated single layers of VO\(_5\) units having a step of \(\approx 2.9\) Å, with the layers closely related to those occurring in crystalline \(V_2O_5\) [41]. On the other hand, Oka et al. proposed that \(V_2O_5\cdot nH_2O\) is made of \(V_2O_5\) bilayers according to the crystalline structure of \(A_xV_2O_5\) (\(A=Na, K\)) [42]. This model is further supported by polarised X-ray absorption and XAFS studies on deposited \(V_2O_5\cdot nH_2O\) films [43].

The structure of \(V_2O_5\cdot nH_2O\) xerogel has been determined for the first time using the atomic pair distribution function (PDF) technique [18] (see Fig. 9). The structure differs from that of crystalline \(V_2O_5\) and is consistent with the one proposed by Oka et al. At the atomic scale, the xerogel can be well described as a double layer of \(V_2O_5\) stacked along the \(c\)-axis of a monoclinic unit cell. The stacking sequence is imperfect, confirming the widespread view of the extensive turbostatic disorder in this system. The closest distance between the bilayers is about 11.5 Å. When the xerogel intercalates or extracts guest species, this distance expands or contracts correspondingly. While, the distance between the two single sheets of \(V_2O_5\) making up the bilayer slab is approximately 2.90 Å [2] (see Fig. 9).

![Structure of \(V_2O_5\cdot nH_2O\) xerogel as revealed by PDF analysis. Water molecules are shown in green.](image)

### 3. Vanadium pentoxide nanostructures as Lithium battery electrodes

#### 3.1. Nanorod arrays

Synthesis and electrochemical properties of single-crystal \(V_2O_5\) nanorod arrays were first reported by Cao, Wang and
co-workers. They have utilized a template-based electrodeposition method by depositing V$_2$O$_5$ into pores of polycarbonate (PC) templates, with the assistance of an electric field from three different types of solution (see Fig. 10). The formation of single-crystal nanorods from solutions by electrochemical deposition is attributed to evolution selection growth (Scheme 1 (a)). While, in the case of nanorods made from the V$_2$O$_5$ sol by electrophoretic deposition, the formation of single-crystal nanorods is explained by homoepitaxial aggregation of crystalline nanoparticles (Scheme 1 (b)).

Patrissi and Martin deposited triisoproxyvanadium (V) oxide into the pores of polycarbonate using the template synthesis by removal of membranes at high temperature to form orthorhombic V$_2$O$_5$ nanorod arrays [16]. They found that the arrays deliver three times the capacity of the thin film electrode. After that, Li et al. achieved improved volumetric energy densities of V$_2$O$_5$ nanorod arrays by chemically etching the polycarbonate membrane to increase its porosity prior to template synthesis [45]. Sides and Martin prepared V$_2$O$_5$ nanorods of different diameters at low temperature to compare their electrochemical properties. They investigate that the V$_2$O$_5$ nanorods with a diameter of 70 nm deliver higher specific discharge capacities than those with micrometer-sized diameters [46]. Fig. 11 shows the comparison between the current density and lithium ion intercalation capacity of nanorod arrays and sol-gel films measured by chronopotentiograms (CPs).

Fig. 11. Discharge capacity versus current density for V$_2$O$_5$ nanorod arrays and sol-gel derived films, taken from [44].

It is shown that for a given lithium ion intercalation capacity, nanorod arrays from solution route possess an up to five times larger current density than that of sol-gel films, which is larger than that of the sol electrophoresis nanorod. Similarly for a given current density, such as 0.7 Ag$^{-1}$, nanorod arrays can store up to five times more Li than in sol-gel films and more than in the sol electrophoresis nanorod. The differences in their electrochemical properties are attributed to the differences in the size of the structures. Nanorods grown by electrochemical deposition from solutions are dense single crystals, with layers parallel to the nanorod axis. This structure is favourable to lithium ion intercalation and extraction, because the surface oxidation and reduction reactions occur along the surface of the nanorods and the solid-state diffusion distance is very small [14]. In addition, such structure permits the most freedom for dimensional change that accompanies intercalation and extraction reactions. The nanorod grown from sol electrophoresis is also single-crystalline and well-aligned, but it has many defects inside the crystal, which may cause the difference between these nanorod array electrodes.

Fig. 12. (a) Top- and (b) side-view SEM images of electrochemically deposited V$_2$O$_5$ nanotubes within 200 nm diameter pores of a polycarbonate membrane after the membrane was dissolved away in methylene chloride, taken from [48].

Fig. 13. Dependence of the discharge capacity on the cycle number obtained from chronopotentiometric measurements at a cut off voltage of 0.4 to −1.5 V versus Ag/Ag, taken from [49].

3.2. Nanotube arrays

The first successful approach to make vanadium oxide nanotubes was carried out using the carbon nanotubes as a template. It was possible by exploitation of surface-tension effects to coat the carbon nanotubes externally with crystalline layers of a V$_2$O$_5$-like structure [47]. Nanotube arrays of V$_2$O$_5$-nH$_2$O have also been prepared through the template–based electrodeposition method from [VO$_2$]$^+$ solution by using lower voltage and shorter depoosition time compared to the conditions for preparing nanorod arrays [48]. Upon completion of deposition, the sample was dried at 110 °C for 6 h and was then attached to an indium tin oxide (ITO) substrate with silver paste (Ted Pella Inc.). The as-prepared sample on the ITO substrate was dried at 110 °C for another 6 h and was immersed in methylene chloride to dissolve the polycarbonate membrane, resulting in high porosity. It was possible by exploitation of surface-tension effects to coat the carbon nanotubes externally with crystalline layers of a V$_2$O$_5$-like structure [47]. Nanotube arrays of V$_2$O$_5$-nH$_2$O have also been prepared through the template–based electrodeposition method from [VO$_2$]$^+$ solution by using lower voltage and shorter deposition time compared to the conditions for preparing nanotube arrays [48]. Upon completion of deposition, the sample was dried at 110 °C for 6 h and was then attached to an indium tin oxide (ITO) substrate with silver paste (Ted Pella Inc.). The as-prepared sample on the ITO substrate was dried at 110 °C for another 6 h and was immersed in methylene chloride to dissolve the polycarbonate membrane, resulting in high porosity.
in an ensemble of V$_2$O$_5$•nH$_2$O nanotubes on an ITO substrate. Fig. 12 shows top- and side-view SEM images of a V$_2$O$_5$ nanotube array grown within the pores of a polycarbonate membrane after the membrane was dissolved away in methylene chloride.

The V$_2$O$_5$•nH$_2$O nanotube arrays demonstrate an initial high capacity of 300 mAh g$^{-1}$, about twice the initial capacity of 140 mAh g$^{-1}$ from the V$_2$O$_5$•nH$_2$O film (see Fig. 13).

Such enhancement of capacity is due to the large surface area and short diffusion distances offered by the nanotube array [14]. However, degradation was observed during the electrochemical characterization of the arrays, which is slower after the third cycle. In addition, the capacity stabilizes at 160 mAh g$^{-1}$ after the sixth cycle, which is about 30 % higher than the stabilized capacity of the V$_2$O$_5$•nH$_2$O film. The initial degradation of the V$_2$O$_5$•nH$_2$O film suggests that V$_2$O$_5$•nH$_2$O itself prepared from electrochemical deposition has some drawback and suffers a slight loss in electroactivity during cycling. However, the nanotube array shows a more drastic decay of initial performance compared to the film during cycling, possibly due to the morphological flexibility and fragility of the nanotubes, which has also been speculated in the literature [49].

3.3. Nanowires

Pan et al. used a hydrothermal method to synthesize long beltlike nanowires, which are several tens of micrometers long and a few tens of nanometers wide [50]. A hydrothermal method was also later used by Schlecht et al. to synthesize new types of vanadium oxide belts exhibiting a boomerang shape [51]. A layered structure was revealed closely corresponding to vanadium pentoxide nanofibers. The structure of nanobelts is unique in that it originates from twinning along the [130] direction, which is the first observation of twins within individual nanosized crystals. However, the intercalation properties of these beltlike nanowires or boomerang-shaped nanobelts of vanadium oxides were not further investigated. Chang et al. recently reported growth conditions of V$_2$O$_5$ nanowires and demonstrated a simple but noble percolation system composed of the conducting nanowires and insulating matrix [52]. They are prepared from ammonium metavanadate with acidic ion-exchange resin. The length of V$_2$O$_5$ is controlled by keeping the mixture of ammonium metavanadate and acidic ion-exchange resin for a certain time period. Longer aging time leads to longer V$_2$O$_5$ nanowires in the sol, for example, nanowires of a few micrometers long are obtained by keeping the mixture for 20 days. This procedure produced a brown sheet, which is rinsed with water and then peeled off the filter paper, resulting in a free-standing sheet composed of entangled V$_2$O$_5$ nanowires. Gu et al. also demonstrated the growth of V$_2$O$_5$ nanowires using a method similar with Chang et al. for electrochemical actuators [53].

Furthermore, ultralong hierarchical vanadium oxide nanowires can be formed using the following procedure: as NH$_4$VO$_3$/PVA solution jet accelerated toward the grounded aluminium foil, then NH$_4$VO$_3$ crystals began to grow accompanied by rapid evaporation of solvent. Meanwhile, the PVA nanowire played the role of a template for the oriented growth of attached NH$_4$VO$_3$ nanorods, which resulted in the formation of ultralong hierarchical NH$_4$VO$_3$/PVA composite nanowires, as illustrated in Scheme 2 a. Finally, the ultralong hierarchical vanadium oxide nanowires made up of attached nanorods were formed after annealing, as illustrated in Scheme 2 b.

This novel nanostructure exhibits high performance for lithium ion batteries, providing a high discharge capacity of 390 mAh g$^{-1}$ and improved cycling stability, which results from reduced self-aggregation of the nanomaterials [54].

3.4. Nanorolls and nanobelts

Pioneering work on the synthesis and electrochemical properties of vanadium oxide nanorolls has been carried out by Spahr and co-workers [55]. In their synthesis, a combination of sol-gel reaction and hydrothermal treatment
of vanadium oxide precursor is conducted in the presence of structure-directing template. The resultant nanoroll is either constructed in closed concentric cylinders (nanotubes) or formed by scrolling one or more layers (nanoscrolls). If amine is replaced by ammonia during the hydrolysis step, a new type of vanadium oxide nanoroll (nanotube) with alternating interlayer distances is yielded [56]. Such a unique structure is first observed in a tubular phase. Compared to other tubular systems, the vanadium oxide nanorolls are especially interesting because they possess four different contact regions, that is, tube opening, outer surface, inner surface, and interstitial region.

Fig. 15. Cyclic voltammograms for Na-exchanged vanadia nanorolls (30 min exchange time) a) well-ordered vanadium oxide nanorolls, and b) defect-rich nanorolls. The well-ordered material shows many steps in the CV curves, similar to crystalline layered vanadates. By contrast, the CV curves for the defect-rich nanorolls are quite similar to sol-gel prepared vanadia, taken from [57].

The relationship between the nanoscale structure and the electrochemical properties of vanadium oxide nanorolls has been demonstrated by Sun et al. [57]. The well-ordered nanorolls behave similarly to classic crystalline vanadium oxides, while the defect-rich materials behave more closely to sol-gel vanadium oxides, which are only partly crystalline on the atomic scale (see Fig. 15). The combination of nanoscale and atomic scale disorder appears to produce the highly desirable combination of increased specific capacity and better cyclability. It is proposed that the increased capacity is associated with additional redox sites that result from the atomic scale disorder. While the better cycle lifetimes are associated with increased accessibility of alkali ions to the gallery regions between the layers because of the cracks and exfoliations in the defect-rich nanotubes.

Li et al. have studied the synthesis and electrochemical behaviour of orthorhombic single-crystalline $\text{V}_2\text{O}_5$ nanobelts [58]. The $\text{V}_2\text{O}_5$ nanobelts with widths of 100–300 nm, thicknesses of 30–40 nm and lengths up to tens of micrometers are obtained by hydrothermal treatment of aqueous solutions of $\text{V}_2\text{O}_5$ and $\text{H}_2\text{O}_2$ (see Fig. 16 A). In Fig. 16 B and C, it is found that the belts can be cleaved along the (100) and (001) planes as indicated by the arrows according to Fig. 16 E. The belt-like shape of $\text{V}_2\text{O}_5$ is further confirmed by a low-magnification TEM image (see Fig. 16 D). The selected area electron diffraction (SAED) pattern (inset to Fig. 16 D) taken from an individual nanobelt is indexed to orthorhombic $\text{V}_2\text{O}_5$, indicating that a nanobelt is a single crystal, with a preferential growth direction along the [010]. The HRTEM image (see Fig. 16 E) with clear interplanar spacing $d_{100} = 1.78$ Å and $d_{200} = 5.75$ Å shows that the belt is structurally perfect and also conforms the growth direction.

Fig. 16. Taken from [59], SEM (A-C) and TEM (D and E) images of $\text{V}_2\text{O}_5$ single crystalline nanobelts synthesized at 180 °C for 48 h. The insets to panels A and D show the rectangular cross-section of the belt-like materials and the SAED pattern taken from an individual nanobelt respectively (scale bar = 100 nm).
The discharge curves of the commercial V_2O_5 powders and the as-synthesized V_2O_5 nanobelts are illustrated in Fig. 17. In the first circle, the specific discharge capacity of V_2O_5 nanobelts is as high as 288 mAh g\(^{-1}\) in the potential range of 1.5–4.0 V, which is much higher than that of commercial V_2O_5 powders (132 mAh g\(^{-1}\)) and comparable to that reported by Wan’s group (286.4 mAh g\(^{-1}\)) [60]. The higher capacity of V_2O_5 nanobelts may be ascribed to their larger specific surface area [59].

![Discharge curves of the commercial V_2O_5 powders and the as-synthesized V_2O_5 nanobelts.](image-url)

**Fig. 17.** Taken from [59]. Discharge curves of the commercial V_2O_5 powders and the as-synthesized V_2O_5 nanobelts.

V_2O_5+0.6H_2O nanobelts with scroll-like structures were successfully synthesized on a large scale by a simple hydrothermal growth method using NH_4VO_3 as the raw material in the presence of sulfuric acid and acetic acid respectively as shown in **Scheme 3**.

In the formation process, neither templates nor surfactants were used in the reaction system. Based on the experimental results of the hydrolysis and condensation of NH_4VO_3, the H_2O\(^{+}\) intercalating and splitting process was proposed to elucidate the formation of the 1 D nanostructure in which the acid played the determining role for the formation of nanobelts or nanorolls.

The sample obtained with the dilute sulfuric acid to adjust pH, as shown in **Fig. 18 (a), (b)**, is composed of a large number of uniform nanobelts, 100 – 150 nm wide and 20 – 30 nm thick. **Fig. 18 (c), (d)** display the images of the sample obtained in the presence of dilute acetic acid, indicating that the sample consists of uniform straight wire-like nanostructures with diameters of 100 – 250 nm and length of tens of micrometres. From further observation on **Fig. 18 (d)**, these wire-like nanostructures, were scroll-like nanostructures, i.e. nanorolls, and most of them scroll incompletely to form a kind of half-tube as observed in the inset of **Fig. 18 (d)**.

![FE-SEM images of the obtained V_2O_5+xH_2O nanobelts and nanorolls.](image-url)

**Fig. 18.** Taken from [61]. FE-SEM images of the obtained V_2O_5+xH_2O nanobelts and nanorolls. (a), (b) nanobelts obtained using diluted sulfuric acid to adjust pH = 4; (c), (d) nanorolls obtained using diluted acetic acid to adjust pH = 4.

The charge - discharge capabilities of the V_2O_5+0.9H_2O nanobelts, V_2O_5+0.6H_2O nanorolls and anhydrous V_2O_5 nanostructures were investigated by using the samples as the cathode in a lithium-ion battery as shown in **Fig. 19**. **Fig. 19 (a) and (b)** show a typical charge - discharge curve of crystalline V_2O_5 with a stepwise shape due to the phase change of Li-V_2O_5 during lithium ion intercalation [16]. V_2O_5+0.6H_2O nanorolls exhibit a slightly higher initial discharge capacity (253.6 mAh g\(^{-1}\)) than V_2O_5+0.9H_2O nanobelts (223.9 mAh g\(^{-1}\)), which may be attributed to their different morphologies and water contents. In order to investigate the effect of H_2O in crystals on the electrochemical properties of V_2O_5+0.9H_2O nanobelts and V_2O_5+0.6H_2O nanorolls, the electrochemical properties of the as-obtained anhydrous V_2O_5 nanostructures (sample...
500-1 and sample 500-2) were measured under the same conditions, and their charge–discharge profiles in the initial cycle are shown in Fig. 19 (b). The capacitances of samples 500-1 and 500-2 increased to 287.8 mAh g⁻¹ (sample 500-1) and 307.5 mAh g⁻¹ (sample 500-2) demonstrating the significant effect of H₂O on their electrochemical properties. In addition, their cycle performances are shown in Fig. 19 (c), which illustrates the dependence of the discharge capacity on cycle number for these samples. The sample of V₂O₅·0.9H₂O nanobelts and its post-annealing product (sample 500-1) retain nearly 83 % and 74 %, respectively, of their initial discharge capacities after 15 cycles. The capacitance of the V₂O₅·0.6H₂O nanorolls decays relatively fast initially with cycling tests to retain 61 % of its initial discharge capacity after 7 cycles and reaching a stabilized capacitance about 147.0 mAh g⁻¹; the capacitance of its post-annealing product (sample 500-2) decays to retain 67 % of its initial discharge capacity after 15 cycles. The decreasing discharge capacity of the samples is likely due to partial fracture of their shapes and structural degradation after redox cycles.

3.5. Nanoribbons and nanocables

Cui and co-workers prepared V₂O₅ nanoribbons and investigated the dependence of the electrochemical property on the width and thickness of nanoribbons by studying the chemical, structural, and electrical transformations of V₂O₅ nanoribbons at the nanostructured level [62]. They found that transformation of V₂O₅ into the α-LiᵥV₂O₅ phase takes place within 10 s in thin nanoribbons, and the efficient electronic transport can be maintained to charge α-LiᵥV₂O₅ nanoribbon within less than 5 s. Therefore, it is suggested that Li diffusion constant in nanoribbons is faster than that in bulk materials by three orders of magnitude, leading to a remarkable enhancement in power density [63].

Fig. 19. Taken from [61], (a), (b) Charge–discharge curves in the initial cycle of the as-prepared V₂O₅·nH₂O and anhydrous V₂O₅ nanostructures at a given constant current density of 0.6 mA cm⁻². (c) The cycle performances of capacitances of the V₂O₅·nH₂O and anhydrous V₂O₅ nanostructures.

Fig. 20 compares the electrochemical performance of Ni-V₂O₅·nH₂O nanocable arrays, single-crystal V₂O₅ nanorod arrays, and sol–gel derived V₂O₅ films. Ni-V₂O₅·nH₂O nanocable arrays demonstrate remarkably improved capacity and rate capability in comparison with the other two. The intercalation capacities of both nanorod arrays and sol–gel films decrease rapidly as the current density increases, while nanocable arrays are able to retain the high capacity at high current density (discharge rate), indicating the excellent high-rate performance of nanocable arrays [64]. In addition, as shown in Fig. 20 (c), Ni-V₂O₅·nH₂O nanocable array has significantly higher energy density and power density than the nanorod array and sol–gel film by at least one order of magnitude, which is ascribed to the enhanced surface area and reduced internal resistance.

Fig. 20. Taken from [65], a) Cyclic voltammograms of Ni-V₂O₅·nH₂O nanocable array and V₂O₅ nanorod array using a scan rate of 10 mV s⁻¹. b) Relationship between current density and Li intercalation capacity of Ni-V₂O₅·nH₂O nanocable array, V₂O₅ nanorod array and sol–gel film from chronopotentiometric measurements. c) Ragone plot for Ni-V₂O₅·nH₂O nanocable array, V₂O₅ nanorod array and sol–gel film.
3.6. Gels

Nanocomposites of vanadium oxide aerogel with single-walled carbon nanotubes (SWNT) were prepared by integrating the nanotubes as part of the sol-gel chemistry. The nanocomposite electrodes were prepared by mixing the V$_2$O$_5$-SWNT powder with poly(vinylidene fluoride) in cyclohexane to form slurry, which was then applied to a stainless steel mesh [66]. For comparison, traditional electrodes were prepared using carbon black (CB) powder as the conductive additive. The voltammetric responses for the aerogel immobilized on the sticky carbon electrode are shown in Fig. 21. In the LiClO$_4$/propylene carbonate (PC) electrolyte, a broad and capacitive response is observed, and the intercalation peaks appear to be superimposed upon the capacitive response (Fig. 21a). The background from the sticky carbon electrode is negligible. At high scan rates, the aerogel responds as a conventional capacitor (Fig. 21b). In contrast, when the same V$_2$O$_5$ aerogel is prepared in a conventional electrode structure, as a composite with carbon additive and binder, the characteristic intercalation behaviour for sol-gel derived V$_2$O$_5$ materials is observed (Fig. 21c). Finally, electrochemical measurements which were made using an electrolyte of tetrabutyl ammonium perchlorate (TBAP) dissolved in PC show very different behaviours (Fig. 21d). In this case, the response for the aerogel immobilized on the sticky carbon is significantly less than that obtained in the Li$^+$ electrolyte.

![Fig. 21. Taken from [67], voltammetric response of V$_2$O$_5$ aerogel on sticky carbon in LiClO$_4$/PC electrolyte at (a) 0.1 mVs$^{-1}$ and (b) 5 mVs$^{-1}$. The sticky carbon response is also shown. Voltammetric response of a typical composite electrode at (c) 0.1 mVs$^{-1}$. Voltammetric response of the aerogel on sticky carbon in TBAP electrolyte (d).](image)

The specific capacitance and the lithium capacity for the various samples are shown in Table 1. The lithium capacity value (400 mAh$^{-1}$) is nearly three times larger than the lithium capacity of a traditional positive electrode material such as LiCoO$_2$ [6]. Such large energy density shows the benefit of working with electrochemical systems, which exhibit more than one electron redox reactions. The specific capacitance values range from 960 Fg$^{-1}$ to over 2000 Fg$^{-1}$ depending upon the drying method, supercritical or ambient. The largest specific capacitance, 2300 Fg$^{-1}$, was obtained using cyclohexane as the drying solvent and corresponds to nearly 1500 mFcm$^{-2}$. Such a large magnitude of specific capacitance arises from pseudocapacitive behaviour [68], which is a very different charge storage mechanism than that which occurs with traditional intercalation materials [69].

In addition, the data summarized in Table 1 indicate how the aerogel morphology influences electrochemical properties. Supercritically dried aerogels have high surface areas, but include many nanosized pores of less than 5 nm. These small pores may not be readily accessible to the electrolyte and, in that case, their contribution to the sample capacitance will be nominal. In contrast, samples dried with a low surface tension solvent at ambient pressure tend to have larger pores, in the range of 10 to 30 nm. The significant level of pore volume in combination with larger size pores greatly improves the likelihood that electrolyte penetration occurs throughout the entire particle.

| Drying solvent | Specific capacitance (mg$^{-1}$) | Lithium capacity (mAg$^{-1}$) | Surface area (m$^2$g$^{-1}$) | % porosity | Total pore volume (m$^3$g$^{-1}$) |
|----------------|---------------------------------|-----------------------------|-----------------------------|------------|---------------------------------|
| Supercritical | 1200 | 400 | 900 | 200 | 96 | 0.13 |
| Supercritical | 1200 | 400 | 900 | 200 | 96 | 0.13 |
| Supercritical | 900 | 200 | 400 | 155 | 96 | 0.13 |
| Supercritical | 2000 | 400 | 650 | 155 | 96 | 0.13 |
| Supercritical | 3000 | 400 | 100 | 200 | 96 | 0.13 |

4. Electrochemical properties of vanadium pentoxide nanostructures

Fig. 22 shows the change of transmittance intensity at 700 nm as a function of time when an external voltage of 3.0 V is applied for the V$_2$O$_5$ nanorod array grown by electrophoretic deposition from the sol and the sol-gel derived V$_2$O$_5$ film, respectively.

![Fig. 22. Taken from [70], transmittance intensity of nanorod arrays and sol-gel films subjected to an externally applied electric field as a function of time.](image)

A 30 % reduction was achieved in ca. 50 s in the V$_2$O$_5$ nanorod array; however, 300 s were required in the film. In conclusion, both the extent and the speed of change in the
transmittance intensity of the nanorod array are significantly faster than those of the sol-gel derived film, corroborating with an enhanced electrochemical intercalation process in nanorod arrays due to a large surface area for the surface redox reaction and a short and easy diffusion path for mass and charge transport [14]. When a suitable potential is applied for 30 s for each step, the coloration and bleaching states of V$_2$O$_5$ nanowire-based electrochromic device are pale blue and yellow, respectively (see Fig. 23), and the maximum transparent contrast $\Delta T$ is 37.4% at $\lambda = 415$ nm.

Using the same conditions, the switching speed is measured from time dependence of transmittance responses as shown in Fig. 24. The switching time is found to be about 6 s from colored state (+1 V) to bleached state (-1 V) and 5 s from bleached state (-1 V) to colored state (+1 V).

Fig. 23. Taken from [71], transmittance spectrum of the colored (+1 V) and bleached (-1 V) states of the V$_2$O$_5$ nanowires in the range of 340 – 700 nm.

Fig. 24. Taken from [71], transmittance spectrum of the V$_2$O$_5$ nanowires at 415 nm as a function of time upon potential switching between +1 and -1 V.

**Conclusion**

In this paper, a review on the lithium intercalation properties of vanadium pentoxide nanostructures is presented. The morphology and the structure of nanorod arrays, nanotube arrays, nanowires, anhydrous and hydrous nanorolls and nanobelts, nanoribbons, nanocables and gels have been discussed. Their intercalation properties have been investigated and compared to those of the vanadium pentoxide film electrodes. It is demonstrated that the nanostructured electrodes exhibit enhanced storage capacity and rate performance over those of V$_2$O$_5$ films. In particular, the Ni-V$_2$O$_5$$\cdot$$n$H$_2$O nanocable array presents higher energy density and power density than the nanorod array due to the enhanced surface area and the reduced internal resistance. Further, the electrochemical characterizations of the anhydrous vanadium pentoxide nanorolls and nanobelts have demonstrated higher discharge capacity and more sustainable cycling life than the hydrous nanostructures. Finally, it is shown that both the extent and the speed of change in the transmittance intensity of the nanorod array are significantly faster than those of the sol-gel derived film. There is much chemistry still remaining to fully understand the intercalation properties of vanadium pentoxide nanostructures with respect to lithium and other cations. On the other hand, there is a lot of research work to go related with the opportunities of the nanostructured materials as electrochromics. The vanadium pentoxide nanostructures will continue their ongoing improvement with further decreasing their size offering enhanced electrochemical performance at a low growth cost. However, the low cost of the lithium batteries and the electrochromics will only be achieved if the costs of the other components such as anodes, transparent conducting materials and electrolytes are also reduced. This can possibly be achieved using carbon related materials and solid instead of liquid electrolytes.

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