Fabrication of Ti substrate grain dependent C/TiO₂ composites through carbothermal treatment of anodic TiO₂†

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Composite materials of titania and graphitic carbon, and their optimized synthesis are highly interesting for application in sustainable energy conversion and storage. We report on planar C/TiO₂ composite films that are prepared on a polycrystalline titanium substrate by carbothermal treatment of compact anodic TiO₂ with acetylene. This thin film material allows for the study of functional properties of C/TiO₂ as a function of chemical composition and structure. The chemical and structural properties of the composite on top of individual Ti substrate grains are examined by scanning photoelectron microscopy and micro-Raman spectroscopy. Through comparison of these data with electron backscatter diffraction, it is found that the amount of generated carbon and the grade of anodic film crystallinity correlate with the crystallographic orientation of the Ti substrate grains. On top of Ti grains with ~(0001) orientations the anodic TiO₂ exhibits the highest grade of crystallinity, and the composite contains the highest fraction of graphitic carbon compared to Ti grains with other orientations. This indirect effect of the Ti substrate grain orientation yields new insights into the activity of TiO₂ towards the decomposition of carbon precursors.

1. Introduction

Composite materials of titania and conductive graphitic carbon have gained particular research interest for application in photocatalysis, dye sensitized solar cells (DSSCs), Li-ion batteries and proton exchange membrane fuel cells (PEMFCs) due to the enhanced performance of the composite compared to TiO₂ or C alone.1–5 It was shown that the application-specific performance of C/TiO₂ hybrid materials is highly sensitive to the synthesis route6 and the detailed preparation conditions.1,7,8 There is evidence that the major benefits of functional C/TiO₂ nano-composites, such as their superior stability and electric conductivity, rely on a close contact between conductive carbon and TiO₂.9,10 Hence, an adapted synthesis is of vital importance to optimize application-specific functional properties.

Biphasic C/TiO₂ or mixed TiOₓCᵧ rich in TiC/TiO₂ materials with different morphologies, such as nanotubes, nanoparticles or thin films, can be prepared via carbothermal treatment of initially prepared titania precursors with reactive acetylene (C₂=H₂) in a gas flow reactor.11–15 These materials have been tested as catalyst supports in studies of the electrochemical reduction of oxygen11 or oxidation of small alcohols,12,16,17 and as Li-ion battery anodes,15 and exhibited promising performances. The employed carbothermal route provides realistic process conditions that are relevant for the production of high surface area...
functional materials based on TiO₂ and C. The physico-chemical properties of the thus prepared materials are empirically tailored via the synthesis parameters. An adapted design of functional composites requires a detailed understanding of the correlation between the synthesis conditions and the intrinsic physico-chemical properties of the produced materials.

In this context, planar anodic titania films that are carbothermally treated under reproducible conditions can be used to systematically study the synthesis–property–performance relationships, which are essential to identify the optimal synthesis procedure to obtain specific functional materials via the carbothermal route. Anodic films are particularly interesting due to their tunable morphology, which can be varied from compact films to high aspect ratio self-organized nanotubes. Several studies have been performed on carbothermally reduced compact anodic TiO₂, rich in TiO and TiC phases, to correlate the synthesis conditions with intrinsic physico-chemical properties, such as structure and composition, and with functional performance, such as (electro-)chemical stability or electrical conductivity.

In a previous study, we have shown that the synthesis conditions of TiOₓCᵧ films significantly influence the catalytic activity of thereon deposited Pt nanoparticles towards the electro-oxidation of ethanol. Interestingly, it was found that the electrochemical stability of Pt/TiOₓCᵧ appeared to be affected by the grain orientations of the polycrystalline Ti substrate. This indicates that the texture of the Ti substrate, on which the precursor anodic film is prepared, constitutes an additional parameter that has to be taken into account when optimizing the synthesis of TiOₓCᵧ films.

In view of the importance of C/TiO₂ composites for applications in energy conversion and storage, we present a study on the preparation of a thin carbon film on compact anodic TiO₂ on polycrystalline Ti (C/TiO₂/Ti_poly) by means of carbothermal treatment with C₂H₂ in a flow reactor at 550 °C. A special focus is given to the investigation of the substrate grain dependent physico-chemical properties of the C/TiO₂ composite film.

The films are ex situ characterized by means of local analysis tools. Electron backscatter diffraction (EBSD), micro-Raman spectroscopy, scanning electron microscopy (SEM), atomic force microscopy (AFM) and scanning photoelectron microscopy (SPEM) are used to correlate grain orientations of the polycrystalline Ti substrate with the local crystalline phase composition, surface topography and chemical composition of the C/TiO₂ films.

The present study sheds light on the origin of the substrate grain effect in carburized TiO₂/Ti_poly systems and thereby shows the significance of the TiO₂ precursor properties for the synthesis of optimized functional C/TiO₂ composites via the carbothermal route.

2. Experimental section

Disks of 1 mm thickness and diameters of 10–15 mm were cut from a 20 mm diameter polycrystalline Ti rod (99.6% purity, temper annealed, Advent Ltd, England). One side of the sample was mechanically and electrochemically polished, following the procedure described in ref. This indicates that the texture of the Ti substrate, on which the precursor anodic film is prepared, constitutes an additional parameter that has to be taken into account when optimizing the synthesis of TiOₓCᵧ films.

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The crystallographic texture of the electropolished Ti substrate was mapped by EBSD (see Fig. 1) using a FEI XL30 scanning electron microscope (SEM) operated at 20 kV accelerating voltage and equipped with a TSL-EDAX EBSD system. The step size of the EBSD map was set to 2 µm, which led to a suitable spatial resolution of the microstructure.
The electropolished Ti samples were consecutively cleaned in an ultrasonic bath with ethanol (technical grade), isopropanol (high purity) and de-ionized (DI) water (MilliPore-Milli-Q system, 18.2 MΩ). Compact amorphous TiOx films were produced by potentiostatic electrochemical anodization in a home-built Teflon electrochemical cell with a two-electrode configuration using a DC power supply controlled by a multimeter, and applying the following settings: anodization at 20 V for 600 s in 0.1 M sulfuric acid (H2SO4, analytical grade, 95–97%, Merck, Germany) electrolyte at room temperature using a platinum mesh counter electrode.14 The anodic films were rinsed with DI water and dried in an argon (Ar 4.8, Linde, Germany) stream.

A carbon layer was synthesized on the anodic film by carbothermal treatment in a tubular quartz reactor under controlled gas flow. The gas flow and acetylene dose were optimized for the reactor geometry. In a reactor tube of 40 mm diameter, the following procedure was applied: [i] purged for two hours under a high flow of Ar to remove air, [ii] heated for up to 90 min at a constant rate to 550 °C in 200 standard cubic centimeters per minute (scbm) Ar, [iii] maintained for 60 min at 550 °C, [iv] 0.5 flow% of acetylene (C2H2, solvent-free, Linde, Germany) was added for 5 min, [v] maintained for 60 min at 550 °C in Ar, and [vi] the oven was left to cool down to room temperature. These samples will be referred to as C/TiO2. A reference sample (termed TiO2 ref) was thermally treated in 200 scbm Ar without C2H2, according to the sequence [i]–[ii]–[v]–[vi].

After anodization and thermal treatment, optical micrographs were acquired for the reference sample using a camera of 10× magnification (belonging to the AFM equipment). The surface morphology of TiOx and C/TiOx was investigated using a field emission SEM based on a Gemini column in a Zeiss CrossBeam NVision 40 system. Micrographs were acquired with the in-lens secondary electron detector using an acceleration voltage of 4 kV and working distances of 4–7 mm. AFM was performed as described in the ESL.†

Micro-Raman spectroscopy was applied to investigate the crystalline phase composition of C/TiO2 and TiO2 ref. A Jobin Yvon Horiba HR800 instrument equipped with a Nd:YAG laser (532 nm) was used in a non-focal operation mode and without a polarizer to analyze the average chemical composition and structure of the film on top of individual Ti substrate grains. The size of the focused laser spot on the sample was about 5 µm using the 50× magnification of the optical microscope. Backscattered Raman signals were recorded with a resolution of less than 2 cm−1 (as determined by measuring the Rayleigh line) in a spectral range from 80 to 2000 cm−1.

SPEM was performed at the ESCAmicroscopy beamline at the Elettra Synchrotron Facility in Trieste, where the X-ray photon beam was demagnified by a Zone Plate to a sub-micron spot of about 150 nm onto the sample, which was then rastered to produce an image by detecting the photoelectrons generated from the sample.25 In this work the photon energy was set to 756 eV. The incident X-ray beam was normal to the sample surface while the angle between the hemispherical electron analyzer (HEA) and the sample surface was 30°. Photoelectron maps of 50 × 50 µm2 were recorded by sampling the surface with a step of 0.2 µm and a dwelling time of 60 ms per pixel. The HEA was equipped with a multichannel electron detector which simultaneously acquires 48 maps (channels), each tuned to a specific photoelectron energy within a selected energy window. This allowed the (i) extraction of spectra from a selected area of the acquired photoelectron micrograph with an energy window of 7.8 eV and a step of 0.164 eV and (ii) removal of the topographic contributions to the photoelectron micrographs and extraction of the chemical contrast. The chemical contrast micrographs (chemical maps) were obtained by selecting, from the recorded 48 maps, those acquired at the photoelectron peak energy and those acquired at an energy related to the background of the photoelectron peak, integrating them to obtain two maps corresponding to the photoelectron peak and background intensity, and determining their ratio.22 The ratio of photoelectron maps acquired in the spectral regions of the C 1s and Ti 2p core levels with their corresponding background is referred to as C/BG and TiO2/BG, respectively. For elemental analysis of the C/TiO2 composite, the uncorrected absolute peak area was used as a qualitative measure, which was obtained from integration of the spectra from the image after subtraction of a constant background.

Additional information about the performed experiments can be found in the ESL.†

3. Results

3.1 Identification of substrate grain orientations

The crystallographic orientations of the Ti substrate grains are determined by EBSD, which provides the three Euler angles, ϕ1, ϕ and ϑ2, which define the crystallographic orientation of the hexagonal unit cell with respect to the substrate surface (according to the Bunge convention25). The sketch in Fig. 1 illustrates the orientation of the hexagonal lattice25 with respect to the substrate surface: ϕ gives the tilt angle of the unit cell c-axis with respect to the surface normal and ϑ2 gives the azimuthal rotation of the hexagonal unit cell around its c-axis.24 A rotation of the Ti substrate around its surface normal is described by the angle ϕ1 (not shown in Fig. 1). The two Euler angles ϕ and ϑ2 can be restricted to 0° ≤ ϕ ≤ 90° and 0° ≤ ϑ2 ≤ 30° due to the symmetry of the hexagonal lattice. For the azimuthal rotation the zero point is chosen to coincide with the (1010) plane for ϕ = 90°, while ϑ2 = 30° and ϕ = 90° yield the (2110) plane. The crystallographic orientation of individual substrate grains can be illustrated in a surface-normal projected inverse pole figure orientation map (from now on termed EBSD map) like in Fig. 1c that uses an RGB color code defined in the standard triangle in Fig. 1b. From the colors in the standard triangle the crystallographic direction parallel to the sample surface normal can be deduced.

EBSD and optical microscopy are used to generate a corre lation map between the Ti substrate grain orientation and the optical appearance of the anodic oxide film. Fig. 1 shows the EBSD map of an electropolished titanium substrate and an optical micrograph of the same surface after anodization. According to
ellipsometry measurements (not shown), the fresh anodic TiO$_2$ film has an average thickness of $\sim 53$ nm. The comparison of Fig. 1c and d reveals that the interference colors of the anodic TiO$_2$ are strongly affected by the orientation of the Ti substrate grains. In particular, the substrate tilt angle $\Phi$ has an important impact on the optical properties of the oxide film. Variations in interference properties are directly related to variations in the TiO$_2$ film thickness. Under the applied synthesis conditions, dark colors correspond to thinner films, and bright colors to thicker TiO$_2$ films.$^{26,27}$ We can define four ranges of substrate tilt angles $\Phi$ that cause different interference colors of the oxide film (see Fig. 1b). The azimuthal substrate orientation, described by $\varphi_2$, has only a minor effect on the interference colors of the anodic TiO$_2$ film (see ESI†). Due to the rotational symmetry of the oxide film thickness on individual grains, the corresponding orientation $\varphi_1$ does not affect the interference color of the anodic TiO$_2$. Table 1 summarizes the assignment of the oxide colors to the tilt angles $\Phi$ of the Ti substrate. This allows an estimation of the substrate orientations without the necessity of an EBSD measurement. The two orientations $\sim$Ti\{1010\} ($\varphi_2 \sim 0^\circ$) and $\sim$Ti\{2110\} ($\varphi_2 \sim 30^\circ$) can be deduced from the optical appearance of the anodic film (without EBSD map) from the optical appearance of the anodic film after thermal treatment at 550 °C (see Fig. S1 and the discussion in ESI†).

In the following, we will focus on the relationship of local chemical and physical properties, obtained by microscopic and spectroscopic techniques of planar C/TiO$_2$/Ti$_{poly}$ composite materials, and the tilt angle $\Phi$ of the Ti substrate grains. The restriction to $\Phi$ is motivated by studies of the TiO$_2$/Ti$_{poly}$ system reported in the literature, which suggest that variations of the tilt angle $\Phi$ have a stronger influence on many physical and chemical properties of anodic TiO$_2$ than the c-azimuthal orientation (given by $\varphi_2$).$^{25,28,29}$

### 3.2 Thermal treatment with C$_2$H$_2$

Fig. 2 shows top-view scanning electron micrographs of the as-grown anodic TiO$_2$ and the C/TiO$_2$ composite film in the same surface area. On both films, the grain boundaries of the underlying Ti substrate are clearly visible. The crystallographic orientations of the substrate grains are indicated by surface projected hexagonal cells. The morphology of the anodic film varies for differently oriented substrate grains (Fig. 2b), which is a

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**Table 1** Assignment of crystallographic Ti substrate orientations to anodic TiO$_2$ film colors

| Interference color (thickness) of TiO$_2$ | Dark red (thin TiO$_2$) | Dark violet/blue | Blue | Bright blue (thick TiO$_2$) |
|------------------------------------------|-------------------------|------------------|------|-----------------------------|
| Tilt angle $\Phi$ of Ti                  | $0^\circ \leq \Phi \leq \sim 15^\circ$ | $\sim 15^\circ \leq \Phi \leq \sim 40^\circ$ | $\sim 40^\circ \leq \Phi \leq \sim 50^\circ$ | $\Phi \geq \sim 50^\circ$ (i.e. all $\sim$Ti\{hkil\}) |
| Ti\{hkil\}$^a$                           | $\sim$Ti\{0001\}        |                  |      | $\sim$Ti\{1010\}           |
| EBSD map colors                          | Red                     | Pink-orange      | Purple-yellow | Blue | Green |
| Representative hexagonal cell            | ![Hexagonal Cell]        |                  | ![Hexagonal Cell] | ![Hexagonal Cell] | ![Hexagonal Cell] |

$^a$ See also ESI.

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![Fig. 2](https://example.com/fig2.png) Scanning electron micrographs of a selected area on the anodic TiO$_2$ film (a and b) and of the same area on the C/TiO$_2$ film (c and d), with hexagonal cells representing the Ti substrate grain orientations.
well-studied phenomenon.\textsuperscript{29} After the carbothermal treatment at 550 °C, a granular film has formed on top of the anodic film (Fig. 2d). The morphology and the coverage of the deposit seem to be different on different substrate grains.

The crystalline phases and the chemical composition of the C/TiO\textsubscript{2}/Ti\textsubscript{poly} composite are determined using micro-Raman spectroscopy. The average Raman response of the film on top of single substrate grains is measured, since the focused laser beam has a lateral size of \(~5\) \(\mu\)m, whereas the grains have a size of \(10\)–\(20\) \(\mu\)m. In the present case, the penetration depth is much larger than the C/TiO\textsubscript{2} film thickness (see discussion in the ESI†), therefore the Raman signal originates from the entire composite film. Fig. 3 shows micro-Raman spectra of C/TiO\textsubscript{2} for three ranges of Ti substrate tilt angles \(\Phi\). To exclude contributions of carbon contamination from secondary sources, no EBSD map was obtained for the electropolished Ti surface of this sample. All six bands of first order Raman scattering of anatase TiO\textsubscript{2} are detected between 100 and 830 \(\text{cm}^{-1}\) in the C/TiO\textsubscript{2} film.\textsuperscript{30} Due to the high proximity of the A\textsubscript{1g} band and the high frequency B\textsubscript{1g} band, they appear as one peak at \(~517\) \(\text{cm}^{-1}\).

The weak shoulders at \(320\) \(\text{cm}^{-1}\) and \(695\) \(\text{cm}^{-1}\) can be attributed to combination bands of the anatase spectrum due to second-order Raman scattering.\textsuperscript{30} The two most intense peaks of rutile, \(E\text{\textsubscript{g}}\) and \(A\text{\textsubscript{1g}}\), are present, as well as its most prominent combination band at \(~235\) \(\text{cm}^{-1}\).\textsuperscript{31} Peaks found at \(~267\) \(\text{cm}^{-1}\) and \(~347\) \(\text{cm}^{-1}\) can be attributed to the most intense Raman signals of Ti\textsubscript{3}O\textsubscript{5}.\textsuperscript{32,33} The broad features between \(~240\) and \(~395\) \(\text{cm}^{-1}\) cannot be clearly identified. The Raman spectra reveal that the anodic TiO\textsubscript{2} contains crystalline domains of anatase and rutile and is partially reduced after the carbothermal treatment with C\textsubscript{2}H\textsubscript{2}.

The spectra detected for different ranges of \(\Phi\) clearly show that the phase composition of the film is affected by the underlying Ti substrate grain orientations. The fractions of crystalline anatase and rutile phases in the film can be deduced from the relative intensities of the corresponding Raman bands.\textsuperscript{34} A comparison between the most intense anatase B\textsubscript{1g} band (\(\sim399\) \(\text{cm}^{-1}\)) and the rutile \(E\text{\textsubscript{g}}\) band (\(\sim447\) \(\text{cm}^{-1}\)), and between the rutile A\textsubscript{1g} band (\(\sim612\) \(\text{cm}^{-1}\)) and the high frequency anatase \(E\text{\textsubscript{g}}\) band (\(\sim639\) \(\text{cm}^{-1}\)) indicate that there is considerably more rutile present in the thermally treated anodic TiO\textsubscript{2} film on top of grains with \(\Phi \leq \sim15^\circ\) than in the film on top of grains with \(\Phi \geq \sim50^\circ\). Grains with \(15^\circ \leq \Phi \leq \sim50^\circ\) exhibit mixed phase compositions.

All spectra depicted in Fig. 3 show two strong broad peaks at \(~1348\) and \(~1600\) \(\text{cm}^{-1}\) that are the characteristic D- and G-bands of carbon and can be attributed to nanocrystalline graphite (NGC).\textsuperscript{35} The ratio of the D-band intensity to the G-band intensity is determined by means of a 4-peak Gaussian fit and yields values of 0.7–0.8 independent of the substrate grain orientations (see Fig. S2 in the ESI†), which correspond to NGC clusters of approximately 6 nm diameter according to the Tuinstra–Koenig relation.\textsuperscript{35}

The carbon signal intensity appears to be influenced by the substrate grain orientation. In Fig. 3 the integrated and normalized areas of the Raman signals that correspond to vibrations of carbon, \(I(C)\), are plotted \textit{versus} the tilt angles \(\Phi\) of the Ti substrate grains. \(I(C)\) decreases markedly with increasing substrate tilt angle; on top of substrate grains with \(\Phi \leq \sim15^\circ\), the intensity is about 2 times higher than on top of substrate grains with tilt angles \(\Phi \geq \sim50^\circ\). The Raman response of the C/TiO\textsubscript{2} composite film differs considerably from the spectrum of a reference sample (TiO\textsubscript{2}\textsuperscript{ref}) that underwent the same thermal treatment without the addition of C\textsubscript{2}H\textsubscript{2} (see Fig. 3). The intensities of the carbon bands detected on TiO\textsubscript{2}\textsuperscript{ref} are much lower than the ones detected on the C/TiO\textsubscript{2} composite, which evidences that the high amount of carbon in the latter originates from the decomposition of C\textsubscript{2}H\textsubscript{2} and not from carbon contamination.

SPM is performed on the C/TiO\textsubscript{2} composite to gain complementary information about the local chemical surface composition and its dependency on the Ti substrate grain orientation. The corresponding results are summarized in Fig. 4. Cross-correlation with EBSD allows for the assignment of crystallographic orientations to the Ti substrate grains underneath the anodic TiO\textsubscript{2} film. It is important to note that only the region at the left of the dashed line in Fig. 4b coincides with the EBSD map that was acquired on the Ti substrate, which enables the identification of possible carbon contamination generated during the EBSD measurement. The grain labeled A in Fig. 4 belongs to substrate orientations with \(0^\circ \leq \Phi \leq \sim15^\circ\), whereas the grains labelled B and C are
characterized by underlying substrate tilt angles of \( \sim 15^\circ \leq \Phi \leq 40^\circ \) and \( 40^\circ \leq \Phi \leq 50^\circ \), respectively. Grains D and E correspond to substrate orientations with \( \Phi \geq 50^\circ \) (see hexagonal cells in Fig. 4a). From Fig. 4 it can be seen that the chemical composition of the carbothermally treated film surface is considerably affected by the properties of the Ti substrate grains. In the chemical map of Fig. 4b brighter contrast corresponds to a relatively higher C 1s signal, evidencing an accordingly higher amount of carbon. In the same areas, the contrast in Fig. 4c appears darker. This can be explained by the presence of a thicker carbon layer on top of the anodic TiO\(_2\) film that attenuates the intensity of the Ti 2p core level photoelectrons. Interestingly, the film on grain C, whose underlying Ti substrate is tilted by \( \Phi = 46^\circ \), appears to be just at the transition from high to low carbon content, since it is barely distinguishable from the surrounding area of the homogeneous shade in the chemical maps. No change in contrast can be observed across the dashed line in Fig. 4b, which proves the absence of artefacts caused by the acquisition of the EBSD map. Fig. 4d and e report the C 1s and Ti 2p\(_{3/2}\) spectra extracted from the indicated areas of the photoelectron micrographs. The C 1s peaks of all the studied grains are clearly dominated by the signal of graphitic carbon (sp\(^2\) hybridized carbon), detected at 284.5 eV. This is in accordance with the Raman response of the C/TiO\(_2\) composite, which proves the presence of NCG. The weak shoulder occurring at 285.6 eV accounts for a small fraction of diamond-like carbon (DLC, sp\(^3\) hybridized carbon). No defined peak of carbide carbon (C–Ti bonds) is found at 281.7 eV (see Fig. S3 in the ESI†), hence only a negligible amount of this species is present at the C/TiO\(_2\) interface. For all studied grains, the Ti 2p\(_{3/2}\) region is dominated by the TiO\(_2\) peak at 459 eV\(^{20}\) that originates from the anodic film. The small shoulder at the low binding energy tail of the peak reveals the presence of reduced titania species, such as Ti(III) and TiC after shorter annealing times and lower temperatures.\(^{20,36}\) However, such TiO and TiC components are treated with acetylene in a flow reactor with the supporting gas argon, which involves higher oxygen partial pressures than under UHV conditions. This is likely that these conditions hinder the formation of TiO\(_2\)C\(_y\) at 550 °C in favor of effective deposition of a carbon layer, or that only a low fraction of TiC and TiO species forms, which are easily re-oxidized upon exposure to air.

Since SPEM yields a photoelectron signal of TiO\(_2\) on all substrate grains, the maximum carbon layer thickness can be estimated from the probing depth for atomically planar graphitic carbon, which yields a thickness of about 2.3 nm. A close look at the chemical maps reveals that the film on top of individual substrate grains has no perfectly homogeneous composition (see black spots on grain A in Fig. 4b), which is supported by the inhomogeneous coverage of the deposited
film that is observed by SEM (Fig. 2d). This finding is most likely related to the properties of the anodic TiO$_2$ at the moment that C$_2$H$_2$ interacts with it during the carbothermal treatment, which is described in the next section.

### 3.3 Thermal treatment without C$_2$H$_2$

During carbothermal treatment, C$_2$H$_2$ is added to the process after one hour of annealing at 550 °C under an argon gas flow. It can be expected that prior to the introduction of C$_2$H$_2$, the properties of the anodic TiO$_2$ film, in particular its crystallinity, are different from those of the pristine film and that these properties affect the interaction with C$_2$H$_2$. Hence, the investigation of the substrate grain-dependent properties of the anodic TiO$_2$ film prior to the addition of C$_2$H$_2$ may unravel the origin of the grain-dependent chemical composition of the final C/TiO$_2$/Ti$_{polycr}$ composite. For this purpose, a reference sample (TiO$_2$$_{ref}$) was prepared under the same conditions as the C/TiO$_2$ composite, but the annealing treatment was stopped prior to the introduction of C$_2$H$_2$. An EBSD map of the polycrystalline Ti substrate recorded after electropolishing allows a precise identification of the tilt angles $\phi$ of the grains.

Fig. 5a shows the Raman response of the TiO$_2$$_{ref}$ film on individual substrate grains that exhibit different tilt angles $\phi$. The colors of the spectra correspond to the substrate grain orientations that are indicated by black spots in the standard triangle (the studied grains are depicted in Fig. 1). Along the arrows, the tilt angle $\phi$ of the substrate grains varies from 0° to 90°. In all spectra the characteristic bands of first-order Raman scattering of crystalline anatase TiO$_2$ are detected. The weak peaks at 320 cm$^{-1}$ and 796 cm$^{-1}$ are combination bands of anatase.$^{30}$ Apart from a weak feature around ~235 cm$^{-1}$, no distinct peaks in the spectrum can be attributed to rutile, which suggests a very low fraction of this phase in the TiO$_2$$_{ref}$ film. All spectra show two peaks at ~1340 and ~1600 cm$^{-1}$ that are the characteristic D- and G-bands of carbon, which indicate the presence of NCG.$^{35}$

Fig. 5b shows the spectra of an anodic TiO$_2$ film on top of two different Ti substrate grains with $\phi \geq \sim 50^\circ$ (i.e. grains belonging to the group of $\sim$Ti(hk0) orientations – see Table 1) and $\phi \leq \sim 15^\circ$ (i.e. $\sim$Ti(0001)). Apart from the weakly protruding $E_g$ peak of anatase TiO$_2$ at 143 cm$^{-1}$, both Raman spectra are characterized by very broad and undefined features below ~1000 cm$^{-1}$, evidencing that the film is mainly amorphous.$^{37-39}$

The intensity of the Raman signal of the TiO$_2$ film on $\sim$Ti(0001) is slightly higher than that of the film on $\sim$Ti(hk0), which indicates that the grain orientation of the Ti substrate influences the properties of the pristine anodic TiO$_2$ film on top. The comparison between the Raman response of the TiO$_2$$_{ref}$ system and the pristine anodic TiO$_2$ film allows the changes that have occurred upon annealing; the thermal treatment at 550 °C leads to the crystallization of the anodic TiO$_2$ film. No graphitic carbon is detected on the pristine anodic TiO$_2$. The signal between 840 and 890 cm$^{-1}$, however, could be assigned to the Raman active C-C and C-O stretching of organic carbon contamination.$^{38-43}$

The carbon detected on TiO$_2$$_{ref}$ may therefore originate from organic contamination adsorbed onto the surface of anodic TiO$_2$ and converted to graphitic carbon during the thermal treatment. An electropolished titanium substrate yields a flat baseline signal with almost no features (black spectrum in Fig. 5b). Only at around 143 cm$^{-1}$ a small peak is detected which can be attributed to the corresponding $E_g$ band of anatase TiO$_2$, arising most likely from the natural oxide film that is usually present on titanium. There is another small feature at about 1555 cm$^{-1}$, which might originate from carbon contamination as well and which is also present on the anodic TiO$_2$ film.

A strong influence of the substrate grain orientation on the Raman response of TiO$_2$$_{ref}$ is found. While the anatase bands that are detected on top of grains with $\phi \leq \sim 15^\circ$ (red spectra in Fig. 5a) are relatively intense and sharp, the same peaks are not only weaker and broader when detected on grains with $\phi \geq \sim 50^\circ$ (green and blue spectra in Fig. 5a), but also embedded in a plateau-like background signal, which can be attributed to a considerable fraction of amorphous TiO$_2$ in these films. Well-defined, intense Raman bands of narrow full width at half maximum (FWHM) are characteristic of an extended single crystal (see Fig. S4 in the ESI)$^{30,44}$ and are hence an indication of high crystallinity of the detected anatase phase. The more
intense the characteristic bands of anatase are, the higher is the fraction of this crystalline phase in the probed volume of TiO$_2$. Peaks of small FWHM indicate a high long-range order and thus sufficiently large crystalline domains. The features of the amorphous TiO$_2$ background are also present, but less pronounced, on grains with intermediate tilt angles, $\sim 15^\circ \leq \Phi \leq \sim 50^\circ$ (orange, yellow, light and dark purple spectra in Fig. 4a), suggesting that the anodic TiO$_2$ film is partially crystalline on these grains. The crystallinity of the anatase phase gradually improves with decreasing $\Phi$, as can be inferred from the evolution of the E$_g$ band at 143 cm$^{-1}$ (inset in Fig. 5a). A close comparison between the Raman spectra of grains 8, 2 and 5b (red in Fig. 5a) reveals a variation in relative peak heights of the three characteristic anatase bands detected between 350 and 650 cm$^{-1}$. This can be attributed to different preferential orientations of the anatase crystallites on these grains and is not related to varying crystallinity [see Fig. S4 and S5 and the discussion in the ESI†].

Together with the improvement of crystallinity in the TiO$_2$ film, an increase in the intensity of the characteristic D- and G-bands of NCG is observed with decreasing $\Phi$. This suggests that the amount of produced graphitic carbon depends on the crystallinity of anodic TiO$_2$, and in particular on its anatase content.

For a detailed analysis of the dependency of the TiO$_2$ crystallinity and the carbon content on the substrate tilt angle, the low frequency E$_g$ ($\sim 143$ cm$^{-1}$), the B$_{1g}$ ($\sim 396$ cm$^{-1}$) and the high frequency E$_g$ ($\sim 639$ cm$^{-1}$) bands of anatase, as well as the Raman signal of carbon are fitted using Lorentzian and Gaussian line shapes [see Fig. S6 in the ESI†]. The fractions of the crystalline TiO$_2$, the overall (i.e. crystalline and amorphous) TiO$_2$ and the overall carbon phase in the film are then proportional to the integrated areas (i.e. intensities I) of the corresponding Raman signals (see ESI†). To eliminate the possible effects of surface roughness of the TiO$_2$ film on the quantity of generated NCG, the carbon signal intensities are corrected for the topographic surface area ($I_{\text{spec(C)}}$), which was determined by AFM [see Fig. S7 in the ESI†]. The obtained results are reported in Fig. 6.

The overall fraction of crystalline anatase stays almost constant and low for $\Phi \geq \sim 50^\circ$ and abruptly increases below $\Phi \sim 50^\circ$ to five times higher values (Fig. 6a). The intensities of the B$_{1g}$ ($\sim 396$ cm$^{-1}$) and the high frequency E$_g$ bands ($\sim 639$ cm$^{-1}$) of anatase follow the same trends [see Fig. S8a in the ESI†]. In addition, a progressive blue-shift of the B$_{1g}$ band from $\sim 397$ to $\sim 402$ cm$^{-1}$ and a parallel red-shift of the E$_g$ band from $\sim 637$ to $\sim 628$ cm$^{-1}$ are observed when the tilt angle $\Phi$ increases from 0° to 90°, which can be attributed to an increasing fraction of amorphous TiO$_2$ [see Fig. S8b and c in the ESI†].

The FWHM of the anatase E$_g$ band ($\sim 143$ cm$^{-1}$) drops linearly with decreasing $\Phi$ (Fig. 6b). For anatase TiO$_2$ nanocrystals, the FWHM of the E$_g$ band was reported to be inversely related to their dimensions.$^{44,46}$ Hence, this trend indicates a continuous increase of the size of the crystalline anatase domains with decreasing $\Phi$. However, the smallest FWHM value of about 12 cm$^{-1}$ is still higher than the value of 7 cm$^{-1}$ that is expected for bulk anatase TiO$_2$ [see Fig. S4 in the ESI†].$^{44,47}$ Therefore, the thermally treated anodic TiO$_2$ film can be considered polycrystalline on top of every single substrate grain. Similar FWHM values

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**Fig. 6** Evaluation of Raman spectra of TiO$_2$ as a function of the substrate tilt angle $\Phi$. The studied grains are labeled with numbers and marked in the standard triangle. (a) Anatase E$_g$ band intensities at $\sim 143$ cm$^{-1}$ and area-specific carbon signal intensities ($I_{\text{spec(C)}}$); (b) FWHM of the anatase E$_g$ band at $\sim 143$ cm$^{-1}$; (c) The ratio of $I_{\text{spec(C)}}$ and the overall Raman response of crystalline and amorphous TiO$_2$ (I(TiO$_2$)). (d) The ratio of $I_{\text{spec(C)}}$ and the intensity of the anatase E$_g$ band at $\sim 143$ cm$^{-1}$. Dashed lines in (a) and (c): linear least squares fits of carbon band intensities below 55° and above 40°. Dashed line in (d): arithmetic mean (0.85 ± 0.27).
were obtained for anatase TiO₂ nanoparticles with a crystallite size of less than ~20 nm,⁴⁶ thus the domain sizes for thermally treated anodic TiO₂ are in the nanometer range.

The area-specific carbon signal intensities, \( I_{\text{spec}}(C) \), give a qualitative measure of the amount of carbon on the thermally treated anodic film. From Fig. 6a it can be seen that \( I_{\text{spec}}(C) \) follows the same trend with the substrate tilt angle as the intensity of the anatase E\(_g\) band. In Fig. 6c, \( I_{\text{spec}}(C) \) is divided by the overall Raman response of TiO₂, \( I(\text{TiO}_2) \), (including the amorphous phase) on individual substrate grains. As in Fig. 6a, two regions of different slopes can be identified and the crossing of the corresponding least squares linear fits (dashed lines) yields the tilt angle \( \Phi \sim 50^\circ \) as a threshold value. Below this point, \( I_{\text{spec}}(C)/I(\text{TiO}_2) \) and therewith the amount of NCG that is produced on the TiO\(_2\) film upon thermal treatment is increased substantially. When \( I_{\text{spec}}(C) \) is only divided by the intensity of the anatase E\(_g\) band (143 cm\(^{-1}\)), a plot versus \( \Phi \) yields an almost constant line (Fig. 6d). This evidences that the two quantities, i.e. the amount of NCG and crystalline TiO₂ are closely linked. The intensity ratio of the carbon bands is determined to be 0.7–0.8, independent of the crystallographic orientations of the substrate grains underneath the film, which corresponds to NCG clusters of approximately 6 nm on the overall film (see ESI†).

### 4. Discussion

#### 4.1 Crystallinity of the anodic film

The phase evolution of anodic titania films during isothermal annealing proceeds from initially amorphous TiO₂ to phase pure anatase, over mixed phases of anatase and rutile towards phase pure rutile TiO₂, where the final phase composition depends on the annealing temperature, atmosphere and duration.⁴⁸,⁴⁹ A variation of the tilt angle \( \Phi \) of the substrate grains from ~50° to 0° appears to have the same effect on the final crystallinity of TiO\(_2\) as an enhancement of annealing temperature. This implies that the substrate grain orientation, and in particular the tilt angle \( \Phi \), has a strong impact on the kinetics of the phase transition. From the literature it is known that the compact anodic TiO₂ on top of ~Ti[0001] is thinner than on ~Ti[0001] and contains higher donor concentrations, i.e. oxygen vacancies.⁵⁰ This is likely to affect the atomic reorganization or diffusion processes during annealing and therewith yield a higher isothermal crystallization rate for TiO₂ on ~Ti[0001] than for TiO₂ on ~Ti[0001]. In particular, a considerable increase of donor density, supported by a parallel increase in the electron transfer rate, oxygen evolution current and photocurrent, has been found for anodic TiO₂ films on top of Ti substrate grains with tilt angles decreasing below \( \Phi \sim 45^\circ \),⁵¹,⁵² which is in very good agreement with the observed increase of TiO₂ crystallinity for tilt angles \( \Phi \leq \sim 50^\circ \) (see Fig. 6a).

The difference in phase composition of TiO\(_2\) ref and C/TiO₂ is related to the annealing time, which is longer for the thermal treatment with C\(_2\)H\(_2\), hence yielding a considerable fraction of rutile TiO\(_2\) in addition, the carbon present on the TiO₂ may have an effect on the crystallization kinetics.⁵¹,⁵²

#### 4.2 Carbon content in the C/TiO₂ composite

SPERM, micro-Raman spectroscopy and SEM evidence that carbothermal treatment of a compact anodic TiO₂ film with C\(_2\)H\(_2\) at 550 °C yields a C/TiO₂ composite material, consisting of a few nm thick nanocrystalline graphite film on top of slightly reduced, polycrystalline TiO₂. On the reference sample TiO\(_2\) ref which is thermally treated without C\(_2\)H\(_2\), a significantly less amount of carbon is detected, which originates from organic or hydrocarbon contamination of the anodic TiO₂ only.

The carbon signal intensity, extracted from Sperm and micro-Raman spectroscopy, follows a very similar trend with \( \Phi \) for both TiO\(_2\) ref and C/TiO₂. There are mainly two classes of grains: class one comprises substrate grains with tilt angles of 0° ≤ \( \Phi \leq \approx 40^\circ \) where a relatively high amount of carbon is deposited on the oxide film; class two covers all grains with tilt angles \( \Phi \geq \approx 50^\circ \) that exhibit a relatively low fraction of carbon after the carbothermal treatment of the anodic TiO₂. Grains with ~40° ≤ \( \Phi \leq \approx 50^\circ \) belong to a transition zone, where an orientation of \( \Phi \sim 50^\circ \) can be identified as a threshold value.

The nature of the carbon film found on TiO\(_2\) ref and C/TiO₂ is very similar, as evidenced by the resembling shapes of the Raman responses of carbon: the evaluation of the carbon bands' intensity ratios yields a NCG cluster size of approximately 6 nm, independent of the substrate grain orientation and the carbon source.

The formation of a grain-dependent carbon layer under the applied carbothermal treatment strongly suggests that C\(_2\)H\(_2\) (and even organic contamination) is not thermally but catalytically decomposed at the surface of TiO₂.⁵³ Furthermore, the presented results evidence that crystalline TiO₂, in particular the anatase phase, is able to decompose C\(_2\)H\(_2\) since only a small amount of carbon is found on TiO₂ that is mainly amorphous. It has been reported that Ti\(^{4+}\)–O\(^2−\) Lewis acid–base pairs on the surface of crystalline TiO₂ are the active centers for a heterolytic dissociation of C\(_2\)H\(_2\) initiating self-assembly processes of acetylene to form polycyclic aromatic hydrocarbons at room temperature and graphitic carbon at sufficiently high temperatures (650 °C in ref. 53).²⁴,⁵⁵ Anatase TiO₂ was found to be particularly efficient for the decomposition of acetylene, which is in line with our findings.

The crystallinity of compact anodic films on polycrystalline Ti substrates is affected by the Ti substrate grain orientations. This causes the observed grain dependent average chemical surface composition of the obtained C/TiO₂ composite. In a previous in situ study on the conversion of anodic TiO₂/Ti(poly under UHV conditions, a substrate grain dependent chemical composition of the final TiO\(_x\)C\(_y\)/Ti(poly film was found, evidencing that also under different synthesis conditions (temperature, pressure, and carbon source), effects of the polycrystalline substrate need to be taken into account.²⁰ A detailed investigation of the temperature-dependence of the surface chemistry and the structure of this system is ongoing and will be published separately.

From the FWHM values of the anatase E\(_g\) (143 cm\(^{-1}\)) bands it is deduced that the TiO\(_2\) ref film consists of polycrystalline anatase with nanometer size crystallites. This may explain why
only nanocrystalline and no extended graphitic carbon layers are formed in the course of the thermal treatment. Furthermore, it gives an explanation for the inhomogeneous carbon coverage of TiO$_2$ on individual substrate grains as observed with SPEM and SEM, which is particularly emphasized on top of grains that are tilted by $\Phi \leq \sim 15^\circ$: a polycrystalline TiO$_2$ film consists of anatase domains that have different orientations exposing different faces at the surface, which in turn affects their activity towards C$_2$H$_2$ decomposition. The reason for this is that different anatase faces exhibit different fractions of active Ti$^{4+}$--O$^{2-}$ centers. According to the literature a high fraction of these sites appears to be present on dehydroxylated, stoichiometric and extended (001) and/or (010) planes of crystalline anatase.$^{55,56}$

Our results demonstrate that the peculiar properties of compact anodic TiO$_2$ films on Ti$_{\text{poly}}$ have an important impact on the synthesis of C/TiO$_2$/Ti$_{\text{poly}}$ composites via the carbothermal route. The obtained substrate grain dependent chemical composition and structure of the film is likely to affect its overall performance when applied in electrocatalysis or Li-ion battery studies. For example, it is known that local variations in the charge transfer kinetics play a crucial role in the overall activity of electrocatalysts.$^{57}$ The grain dependent electrochemical stability of Pt/TiO$_2$,$^{C_4}$ found in ref. 16 is another consequence of these properties of anodic TiO$_2$ on Ti$_{\text{poly}}$. An adapted design of C/TiO$_2$ films requires a detailed knowledge of the substrate texture and a controlled crystallization of the anodic film during the carbothermal treatment. The presence of crystalline anatase TiO$_2$ required for the reactive decomposition of C$_2$H$_2$ at 550 °C and atmospheric pressure has implications on the carbothermal synthesis of functional C/TiO$_2$ materials in general.

5. Conclusions

In the present work the substrate grain dependent properties of planar C/TiO$_2$/Ti$_{\text{poly}}$ composite films that are synthesized via carbothermal treatment of compact anodic TiO$_2$ with C$_2$H$_2$ in a flow reactor at 550 °C have been investigated. A pronounced correlation between the amount of deposited carbon and the Ti substrate grain orientations is observed: a significantly high amount of graphitic carbon is deposited on the anodic TiO$_2$ film on top of $\sim$Ti(0001) grains than on TiO$_2$ on top of $\sim$Ti{hkl0} grains. The origin of the grain effect is identified as a substrate grain-dependent crystallization of the initially amorphous anodic TiO$_2$ film to nanocrystalline anatase during the thermal treatment, which in turn affects the activity of TiO$_2$ towards C$_2$H$_2$ decomposition to form nanocrystalline graphite. This effect emphasizes the importance of TiO$_2$ precursor crystallinity for the synthesis of C/TiO$_2$ functional materials via a carbothermal route. In the case of anodic TiO$_2$ films, the crystallization needs to be guided towards extended single crystalline anatase domains exposing their most active facets to enable the deposition of extended graphite layers, which, in combination with the possibility of nanostructuring anodic TiO$_2$, provides a highly interesting functional material. Indeed, nanotubular C/TiO$_{2-x}$ composite layers, synthesized via the carbothermal route, have shown promising Li storage capacities,$^{15}$ which may be further improved by means of an optimized synthesis, taking into account the crystallization characteristics of anodic films.

A detailed knowledge of local physico-chemical properties, as-obtained in this study, is important to understand the overall performance of functional materials. In view of the potential applications of C/TiO$_2$ composite materials in energy conversion and storage, the herein introduced planar C/TiO$_2$/Ti$_{\text{poly}}$ composite is proposed as a model system for the investigation of property-performance relationships. Taking advantage of the substrate grain effect, two-dimensional property maps of the C/TiO$_2$ film that correlate (substrate grain dependent) intrinsic physico-chemical properties, such as chemical composition or morphology, and functional properties, such as electric conductivity, catalytic activity or (electro-)chemical stability, will give valuable information about the material. This can be realized by means of local analysis tools, such as SPEM$^{52}$ and micro-electrochemistry.$^{58}$

Author contributions

The manuscript was written through contributions of all authors. CR planned and executed or coordinated the experiments, evaluated the data and drafted the manuscript. MF participated in the SPEM experiments and contributed to the SPEM data evaluation. CVV and LC participated in the SPEM experiments. NB evaluated the EBSD measurements which were performed by SJ. CH performed the Raman measurements. LG and MA gave technical support at the ESCAmicroscopy beamline of the Elettra Synchrotron in Trieste. SA conceived of the SPEM study and gave advice for the data evaluation. GG gave general advisory support. JKL supervised the study and actively participated in drafting the manuscript.

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Conflicts of interest

The authors declare no competing financial interest.

Abbreviations

| Abbreviation | Description                      |
|--------------|----------------------------------|
| UHV          | Ultra-high vacuum                |
| EBSD         | Electron backscatter diffraction |
| AFM          | Atomic force microscopy          |
| SEM          | Scanning electron microscopy     |
| SPEM         | Scanning photodetection microscopy |
| FWHM         | Full width at half maximum       |
| NCG          | Nanocrystalline graphite         |

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