Data-driven exploration of new pressure-induced superconductivity in PbBi$_2$Te$_4$

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ABSTRACT
Candidate compounds for new thermoelectric and superconducting materials, which have narrow band gap and flat bands near band edges, were exhaustively searched by the high-throughput first-principles calculation from an inorganic materials database named AtomWork. We focused on PbBi$_2$Te$_4$, which has the similar electronic band structure and the same crystal structure with those of a pressure-induced superconductor SnBi$_2$Se$_4$ explored by the same data-driven approach. The PbBi$_2$Te$_4$ was successfully synthesized as single crystals using a melt and slow cooling method. The core level X-ray photoelectron spectroscopy analysis revealed Pb$^{2+}$, Bi$^{3+}$ and Te$^{2-}$ valence states in PbBi$_2$Te$_4$. The thermoelectric properties of the PbBi$_2$Te$_4$ sample were measured under high pressure using a diamond anvil cell with boron-doped diamond electrodes. The resistance decreased with increasing of the pressure, and pressure-induced superconducting transitions were discovered at 2.5 K under 10 GPa. The maximum superconducting transition temperature increased up to 8.4 K at 21.7 GPa. The data-driven approach shows promising power to accelerate the discovery of new thermoelectric and superconducting materials.

1. Introduction
A data-driven approach based on high-throughput computation has recently been applied successfully to exploration of new functional materials such as battery materials, thermoelectric materials, superconductors, and so on. Once proper target quantities are selected, this approach may be more efficient than or at least complementary to traditional carpet-bombing type experiments based on experiences and inspirations of researchers [1–5]. We have reported a case study of the

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A data-driven approach through a discovery of pressure-induced superconductivity in a compound SnBi$_2$Se$_4$ selected by the high-throughput screening [6]. In this particular screening, the candidate compounds were explored according to a guideline that is characterized by specific band structures of ‘flat band’ near the Fermi level, such as multivalley [7], pudding mold [8], and topological-type [9] structures. If such kinds of flat band are realized near the Fermi level, thermoelectric properties with high electrical conductivity and Seebeck coefficient would be enhanced [8,10]. If the flat band crosses the Fermi level, superconductivity would be realized due to high density of states (DOS) [11–13]. Experimentally, a single crystal of SnBi$_2$Se$_4$ exhibited an insulator-to-metal transition under 11 GPa [6], in a good agreement with the theoretical prediction. Moreover, a pressure-induced superconductivity was observed with maximum superconducting transition temperature ($T_c$) of 5.4 K under 63 GPa in accordance with our scenario. That work serves as a case study of the important first step for next-generation data-driven material science.

In the aforementioned data-driven approach, a high thermoelectric performance in SnBi$_2$Se$_4$ is expected under high pressure around its insulator to metal transition, since the band gap decreases by the applied pressure, and then the flat band approaches the Fermi level. If a certain compound with the same crystal structure and similar band shape has narrower band gap than that of SnBi$_2$Se$_4$, it will show superior thermoelectric property even at ambient pressure. Furthermore, it could be expected that superconductivity may appear at much lower pressure, compared with SnBi$_2$Se$_4$.

Based on these considerations, we focused on PbBi$_2$Te$_4$ as a target compound because it has the same crystal structure and similar band structure with the band gap narrower than ~200 meV of SnBi$_2$Se$_4$. Superior thermoelectric properties at ambient pressure and the superconductivity under lower pressure could be expected in PbBi$_2$Te$_4$, compared with SnBi$_2$Se$_4$. In this study, we successfully synthesized the sample of PbBi$_2$Te$_4$ in a single crystal. The crystal structure, compositional ratio, and valence states of the PbBi$_2$Te$_4$ single crystal were analyzed by the powder X-ray diffraction (XRD), an energy dispersive X-ray spectrometry (EDX) and an X-ray photoelectron spectroscopy (XPS), respectively. The thermoelectric properties were measured at ambient pressure. The resistivity of the obtained sample was evaluated under high pressure using a diamond anvil cell (DAC) with boron-doped diamond electrodes [14–17].

### 2. Screening procedures in high-throughput first-principles calculations

One thousand five hundred seventy candidates were listed from the inorganic material database named AtomWork [18], based on the following restriction: abundant and nontoxic or less toxic constituent elements, and the number of atoms being less than 16 per primitive unit cell. The candidates were narrowed down by using the restriction of a narrow band gap and high DOS near the Fermi level. By this screening, the number of candidate compounds was reduced to 45. Finally, we checked whether the band gap decreases (or even the metallic behavior appears) under pressure of 10 GPa, and screened out 27 promising compounds. Through the above screening procedures, PbBi$_2$Te$_4$ was chosen as a candidate for new thermoelectric and superconducting materials. The details of our screening scheme in the high-throughput first-principles calculations were given in our previous paper [6].

Figure 1 shows (a) the crystal structure of PbBi$_2$Te$_4$ with trigonal R-3m structure depicted by VESTA [19], (b) the band structure and (c) the total DOS of PbBi$_2$Te$_4$.
obtained by the generalized gradient approximation with spin-orbit coupling. We can see that the band edges of PbBi$_2$Te$_4$ show flat shape near the Fermi level. This feature is quite similar with that of SnBi$_2$Se$_4$. Additionally, the band gap of 101 meV in PbBi$_2$Te$_4$ is less than half of 208 meV in SnBi$_2$Se$_4$. The feature in the band structure of PbBi$_2$Te$_4$ at ambient pressure is similar to that of SnBi$_2$Se$_4$ under pressure of 5–10 GPa [6]. We could expect superior thermoelectric and superconducting properties under relatively low pressure for PbBi$_2$Te$_4$.

3. Experimental procedures

3.1. Sample synthesis

Single crystals of PbBi$_2$Te$_4$ were grown by a melt and slow-cooling method. Starting materials of Pb grains, Bi grains, and Te chips were put into an evacuated quartz tube in the stoichiometric composition of PbBi$_2$Te$_4$. The ampoule was heated at 1000 °C for 1 h, and then slowly cooled down to 800 °C within 20 h in the furnace. After keeping the temperature for 5 h, the ampoule was cooled down to room temperature. The obtained samples were ground and loaded into an evacuated quartz tube again. The sample was heated at 500 °C for 50 h for homogenization of PbBi$_2$Te$_4$ phase.

3.2. Characterization

The crystal structure of the obtained PbBi$_2$Te$_4$ samples was investigated by the powder XRD using the Mini Flex 600 setup (Rigaku) and Cu Kα radiation. The lattice constants were refined using the Conograph software (High Energy Accelerator Research Organization, Japan) [20]. The chemical composition of the sample was evaluated by an EDX analysis using the JSM-6010LA microscope (JEOL). The valence state was estimated by the core level XPS analysis using AXIS-ULTRA DLD (Shimadzu/Kratos) with monochromatic Al Kα X-ray radiation ($hν = 1486.6$ eV), operating under a pressure of the order of $10^{-9}$ Torr. The samples were cleaved using Scotch tape in a vacuum of approximately $10^{-7}$ Torr. The analyzed area was approximately $1 \times 1$ mm$^2$. The binding energy scale was established by referencing the C 1s value of adventitious carbon. The background signals were subtracted by using the active Shirley method implemented in COMPRO software (Surface Analysis Society of Japan, Japan) [21]. The photoelectron peaks were analyzed by the pseudo-Voigt functions peak fitting.

3.3. Transport measurements

Thermoelectric properties, including the electrical resistivity, Seebeck coefficient, and thermal conductivity, were measured by using the thermal transport option (TTO) of physical property measurement system (PPMS/Quantum Design) under ambient pressure from 300 K to 2 K. The power factor and figure of merit were evaluated from the obtained parameters. Resistance measurements of PbBi$_2$Te$_4$ single crystal under high pressure were performed using an originally designed DAC with boron-doped diamond electrodes [14–17]. Figure 2 shows an optical image of the sample space of our DAC. The sample was placed at the center of the bottom anvil where the boron-doped diamond electrodes were fabricated. The undoped diamond insulating layer covers the surface of the bottom anvil except for the sample space and electrical terminal. The details of the cell configuration were described in the literature [16]. Cubic boron nitride powder with ruby manometer was used as a pressure-transmitting medium. The applied pressure values were estimated by the fluorescence from ruby [22] and the Raman spectrum from the culet of top diamond anvil [23] by an inVia Raman Microscope (RENISHAW).

4. Results and discussion

4.1. Crystal structure, composition and valence state

Figure 3 shows a powder XRD pattern of the pulverized PbBi$_2$Te$_4$ single crystal. All observed peaks were well indexed to trigonal $R$-3$m$ structure with lattice constants of $a = b = 4.42$ Å and $c = 41.57$ Å, without any impurity peaks. Here we note that if the sample is synthesized without the annealing process at 500 °C for 50 h which is described in the experimental section, the PbBi$_2$Te$_4$ contamination appears. EDX analysis of the obtained single crystal yields the composition Pb$_{0.8}$Bi$_{2.2}$Te$_{3.8}$ as normalized by Bi, indicating Pb deficiency in the sample. The deficient nature is consistent with a related compound SnBi$_2$Se$_4$ [24].
The valence states of Pb, Bi and Te in PbBi$_2$Te$_4$ were investigated by XPS. Figure 4(a) shows a Pb 4f core-level spectrum of PbBi$_2$Te$_4$. There are two main peaks at 142.4 eV and 137.5 eV corresponding to Pb $4f_{5/2}$ and $4f_{7/2}$ with the valence state of Pb$^{2+}$ [25]. Figure 4(b) shows a Bi 4f core-level spectrum. The Bi 4f photoemission is split into two peaks, one at around 157.8 eV attributed to Bi $4f_{7/2}$ and the other at around 163.1 eV attributed to Bi $4f_{5/2}$ [26]. These main peak positions are corresponding to that of Bi$^{3+}$ valence state. The Te 3d region had two groups of peaks as shown in Figure 4(c), which were observed at 582.8 eV and 572.4 eV, indicating the existence of Te$^2-$, 586.5 eV and 576.1 eV of Te$^{4+}$ due to a surface oxide layer [27]. These Pb$^{2+}$, Bi$^{3+}$ and Te$^{2-}$ are consistent with the formal charge valence of PbBi$_2$Te$_4$. The Pb 4f and Bi 4f spectra contain small peaks at the higher binding energy region which may be due to the surface oxidization or the asymmetric feature of the main peaks [28]. If the asymmetry can be fitted with a Doniach-Sunjic line shape [29], it means the sample is metallic and has high DOS near the Fermi level [30]. Figure S1 in the supplemental materials shows the outcome of peak fitting using the Doniach-Sunjic line shaped pseudo-Voigt functions. The asymmetric parameter $\alpha$ was 0.12 and 0.09 in Bi 4f and Pb 4f, respectively. Although the peak shape is consistent with the flat band model, further investigation is necessary to determine the reason of its asymmetry.

4.2. Thermoelectric properties

Figure 5 shows temperature dependence of the thermoelectric properties including (a) electrical resistivity, (b) Seebeck coefficient, (c) carrier concentration, and (d) thermal conductivity under ambient pressure for PbBi$_2$Te$_4$ (red curves). The results of SnBi$_2$Se$_4$ (blue curves) are also shown for comparison. PbBi$_2$Te$_4$ shows negative slope of resistivity toward lower temperature, namely metallic behavior. The absolute value of resistivity is much smaller than that of SnBi$_2$Se$_4$ [6]. The negative Seebeck coefficient and negative slope of the Hall voltage as a function of applied magnetic field indicate n-type nature of the sample, which may be caused by the excess Bi in the crystal. If the carrier type is tuned from n-type to p-type, the thermoelectric property could be enhanced because the valence band edge provides higher DOS near the Fermi level. A high pressure application, elemental substitution, or...
electric double layer transistor gating are effective for such kind of band tuning. Although the absolute value of the Seebeck coefficient in PbBi₂Te₄ is smaller than that of SnBi₂Se₄, the thermal conductivity is almost same [6]. Consequently, the thermoelectric properties of the power factor and figure of merit were dramatically increased in PbBi₂Te₄ compared to SnBi₂Se₄. The highest values of the power factor ~100 μWm⁻¹K⁻² and figure of merit ~ 0.02 were obtained at 300 K. This tendency of high thermoelectric performance would be originated from the flat band with the small band gap, which is equivalent to the electronic state in SnBi₂Se₄ under high pressure [6].

4.3. In situ resistivity measurement under high pressure

Figure 6(a) shows a temperature dependence of resistance for PbBi₂Te₄ under various pressures from 1.0 GPa to 13.3 GPa. The sample has already exhibited

Figure 5. Temperature dependence of thermoelectric properties in PbBi₂Te₄ and SnBi₂Se₄ under ambient pressure. (a) resistivity, (b) Seebeck coefficient, (c) carrier concentration (inset is a magnetic field dependence of Hall voltage at room temperature), (d) thermal conductivity, (e) power factor, and (f) figure of merit ZT.

Figure 6. Temperature dependence of resistance in PbBi₂Te₄ under various pressures, (a) 1.0–13.3 GPa, (b) 13.3–50.8 GPa.
metallic behavior under ambient pressure as shown in Figure 5(a). The resistance at 1.0 GPa also shows metallic behavior but with a small hump around 200 K. The resistance and the intensity of hump decreased with the increase of the applied pressure. A pressure-induced superconductivity with clear zero resistance was observed under 10 GPa. This critical pressure of the superconductivity is almost half of 20.2 GPa in SnBi$_2$Se$_4$. In this region, the maximum onset transition temperature ($T_c^{\text{onset}}$) and zero-resistance temperature ($T_c^{\text{zero}}$) were 3.4 K and 2.4 K under 13.3 GPa, respectively.

The $T_c$ of PbBi$_2$Te$_4$ suddenly jumped up with the application of further pressure as same as SnBi$_2$Se$_4$. A temperature dependence of resistance from 13.3 GPa to 50.8 GPa is shown in Figure 6(b). The $T_c^{\text{onset}}$ was enhanced from 3.4 K under 13.3 GPa to 8.1 K under 18.0 GPa. In the higher pressure region, the maximum $T_c^{\text{onset}}$ and $T_c^{\text{zero}}$ were 8.4 K and 7.9 K under 21.7 GPa, respectively. The temperature dependences of resistance around the superconducting transitions are summarized in Figure 7. Indeed, the tendency of the $T_c$ increases is quite similar to that of SnBi$_2$Se$_4$ [6]. Both the critical pressures for the lower and higher $T_c$ phases under ~10 GPa and ~20 GPa in PbBi$_2$Te$_4$ are almost half of ~20 GPa and ~40 GPa in the SnBi$_2$Se$_4$, respectively, due to the band gap difference. The higher $T_c$ in PbBi$_2$Te$_4$ compared with that in SnBi$_2$Se$_4$ would be originated from the higher DOS near the Fermi level because the higher pressure application decreases the DOS due to an increase of bandwidth.

Figure 7. Temperature dependence of resistance around superconducting transitions in PbBi$_2$Te$_4$ under various pressures.

Figure 8. Temperature dependence of resistance of PbBi$_2$Te$_4$ in specified magnetic field, under pressure of (a) 13.3 GPa or (b) 21.7 GPa. (c) Temperature dependence of $H_{c2}^{\parallel ab}$ values at 13.3 GPa and 21.7 GPa.
Figure 8 shows temperature dependence of resistance in various magnetic fields under (a) 13.3 GPa, (b) 21.7 GPa. Upper critical field $H_{c2}^//ab(0)$ values were estimated from the Werthamer-Helfand-Hohenberg (WHH) approximation [31] for the Type II superconductor in a dirty limit. A temperature dependence of $H_{c2}^//ab$ values is shown in Figure 8 (c). The $H_{c2}^//ab(0)$ were 2.4 T under 13.3 GPa and 5.9 T under 21.7 GPa.

Figure 9 shows a resistance-pressure phase diagram of PbBi$_2$Te$_4$ single crystal. The resistance of the sample is dramatically decreased by applying pressure. After that, the first superconducting phase was newly discovered under 10.0 GPa. In a higher pressure region above 18.0 GPa, we observed a $T_c$ jump from 3.4 K to 8.4 K. The $T_c$ jump indicates a possibility of appearance of second superconducting phase. Further experimental investigation is required to clear the origin of $T_c$ jump, for example, analysis of pressure distribution in sample space, in situ XRD measurement under high pressures, and theoretical prediction of crystal structure under high pressure. This superconductivity survived up to at least 50.8 GPa. The $T_c$ and $H_{c2}^//ab(0)$ values were almost independent of the applied pressure.

5. Conclusion

Among 27 compounds suggested by the data-driven approach, we focused on PbBi$_2$Te$_4$ from the viewpoint of band similarity to the pressure-induced superconductor SnBi$_2$Se$_4$ which was also chosen by the data-driven approach. The PbBi$_2$Te$_4$ has similar flat band feature with a smaller band gap, compared to that of SnBi$_2$Se$_4$. The synthesized PbBi$_2$Te$_4$ single crystal exhibited better thermoelectric properties at ambient pressure and higher $T_c$ value under high pressure. Especially, the superconductivity occurred under a pressure of approximately 10 GPa lower than ~ 20 GPa in SnBi$_2$Se$_4$, in accordance with our scenario. The present work presents a case study for the important first-step for the next generation data-driven materials science.

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Disclosure statement

No potential conflict of interest was reported by the authors.

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References

[1] Ishikawa T, Oda T, Suzuki N, et al. Review on distorted face-centered cubic phase in yttrium via genetic algorithm. High Pressure Res. 2014;35:37–41.

[2] Ishikawa T, Nakanishi A, Shimizu K, et al. Superconducting $H_2S$ phase in sulfur-hydrogen system under high-pressure. Sci Rep. 2016;6:23160.

[3] Kiyohara S, Oda H, Miyata T, et al. Prediction of interface structures and energies via virtual screening. Sci Adv. 2016;2:e1600746.

[4] Ge Y, Zhang F, Yao Y. First-principles demonstration of superconductivity at 280 K in hydrogen de with low phosphorus substitution. Phys. Rev. B. 2016;93:094525.

[5] Kopnin NB, Heikkilä TT, Volovik GE. High-temperature surface superconductivity in topological flat-band systems. Phys. Rev. B. 2011;83:220503.

[6] Mori K, Usui H, Sakakibara H, et al. Theoretical expectation of large Seebeck effect in PtAs$_2$ and PtP$_2$. J Phys Soc Jpn. 2014;83:031026.

[7] Fries KS, Steinberg S. Fermi-level characteristics of potential chalcogenide superconductors. Chem. Mater. 2018;30:2251–2261.

[8] Ge Y, Zhang F, Yao Y. First-principles demonstration of superconductivity at 280 K in hydrogen sulfide with low phosphorus substitution. Phys. Rev. B. 2016;93:094525.

[9] Matsumoto R, Hou Z, Hara H, et al. Two pressure-induced superconducting transitions in SnBi$_2$Se$_3$ explored by data-driven materials search: new approach to developing novel functional materials including thermoelectric and superconducting materials. Appl Phys Express. 2018;11:093101.

[10] Heinke F, Urban P, Werwein A, et al. Cornucopia of structures in the pseudobinary system (SnSe)$_2$Bi$_2$Se$_3$: a chemical-crystal cryptic. Inorg Chem. 2018;57:4427–4440.

[11] Baghchesara MA, Yousefi R, Cheraghizade M, et al. A simple method to fabricate an NIR detector by PbTe nanowires in a large scale. Matter Res Bull. 2016;77:131–137.

[12] Doniach S, Šunjíć M. Electron-electron singularity in X-ray photoemission and X-ray line spectra from metals. J Phys C. 1970;3:285–291.

[13] Citrin PH, Wertheim GK, Baer Y. Surface-atom X-ray photoemission from clean metals: Cu, Ag, and Au. Phys. Rev. B. 1983;27:3160–3175.

[14] Zatsepin DA, Boukhvalov DW, Gavrilov NV, et al. XPS-and-DFT analyses of the Pb 4f-Zn 3s and Pb 5d-O 2s overlapped ambiguity contributions to the final electronic structure of bulk and thin-film Pb-modulated zincite. Appl. Surf. Sci. 2017;405:129–136.

[15] Konig D, Cha JI, Lai K, et al. Rapid surface oxidation as a source of surface degradation factor for Bi$_2$Se$_3$, ACS Nano. 2011;5:4698.

[16] Matsumoto R, Sasa Y, Fujikawa M, et al. Novel diamond anvil cell for electrical measurements using boron-doped metallic diamond electrodes. Rev Sci Instrum. 2016;87:076103.

[17] Matsumoto R, Hara H, Tanaka H, et al. Pressure-Induced Superconductivity in Sulfur-Doped SnSe Single Crystal Using Boron-Doped Diamond Electrode-Prefabricated Diamond Anvil Cell. J Phys Soc Jpn. 2018;87:124706.

[18] Wang Y, Yamazaki M, Villars P. Inorganic materials database for exploring the nature of material. Jpn J Appl Phys. 2011;50:11RH02.

[19] Momma K, Izumi F. VESTA3 for three-dimensional visualization of crystal, volumetric and morphology data. J. Appl. Crystallogr. 2011;44:1277–1276.

[20] Mori K, Usui H, Sakakibara H, et al. Theoretical expectation of large Seebeck effect in PtAs$_2$ and PtP$_2$. J Phys Soc Jpn. 2014;83:031026.

[21] Heinke F, Urban P, Werwein A, et al. Cornucopia of structures in the pseudobinary system (SnSe)$_2$Bi$_2$Se$_3$: a chemical-crystal cryptic. Inorg Chem. 2018;57:4427–4440.

[22] Baghchesara MA, Yousefi R, Cheraghizade M, et al. A simple method to fabricate an NIR detector by PbTe nanowires in a large scale. Matter Res Bull. 2016;77:131–137.

[23] Zatsepin DA, Boukhvalov DW, Gavrilov NV, et al. XPS-and-DFT analyses of the Pb 4f-Zn 3s and Pb 5d-O 2s overlapped ambiguity contributions to the final electronic structure of bulk and thin-film Pb-modulated zincite. Appl. Surf. Sci. 2017;405:129–136.

[24] Konig D, Cha JI, Lai K, et al. Rapid surface oxidation as a source of surface degradation factor for Bi$_2$Se$_3$, ACS Nano. 2011;5:4698.