Introduction

Organic molecules with pronounced near-infrared (NIR) absorptions are valuable optical, opto-electronic, and bio-applicable materials. However, in spite of all the efforts made to acquire the low-bandgap feature, up to now only a limited number of air-stable closed-shell, neutral organic small molecules have been developed possessing optimal NIR optical activity. Moreover, their absorptions mostly fall in the range of 700–900 nm. Very few neutral organic small molecules possess a desirable extinction ability beyond 1000 nm. The challenges in designing potent NIR dyes lie in the fact that, in order to acquire the low-energy bandgap feature, up to now only a limited number of air-stable closed-shell, neutral organic small molecules have been developed possessing optimal NIR optical activity. Moreover, their absorptions mostly fall in the range of 700–900 nm. Very few neutral organic small molecules possess a desirable extinction ability beyond 1000 nm. The challenges in designing potent NIR dyes lie in the fact that, in order to attain a low-energy bandgap with a significant extinction coefficient, fine tuning the ground and excited electronic states is required to achieve not only suitable energy levels but also an adequate transition dipole moment. Particular care should also be taken to avoid a detrimental effect on the chemo-stability, which could result from over-boosting the HOMO or over-depressing the LUMO.

A common strategy to attain a low-energy bandgap in an organic structure relies on incorporating electron donator (D) and acceptor (A) moieties into the same molecule, preferably linked by a π-conjugated spacer to extend the effective conjugation length. However, in many cases separately installed D and A subunits impart a very limited extinction coefficient to the low-energy excited state. Also, the charge transfer nature may render the optical behavior highly sensitive to environmental polarity variation.

Another effective approach to developing low-bandgap chromophores utilizes large polycyclic π-systems. A major advantage of polycyclic aromatic compounds is their readily tuned HOMO/LUMO energy levels. Through chemical modifications, D/A substituents can also be incorporated. Moreover, large π-systems with delocalized frontier orbitals also promote enhanced light-absorbing ability. Previously, we developed a number of N-hetero-polycyclic dicarboximide molecules manifesting absorptions around 800–900 nm. The syntheses of these molecules harnessed 2,3,6,7-tetrabromo-1,4,5,8-tetraacarboxydiimide (4Br-NDI) as an important synthon. The highly electron-deficient NDI allows exploitation of facile nucleophilic substitution reactions to functionalize and expand the polycyclic π-skeleton. Moreover, the strongly electron-pulling dicarboximide groups help confer an adequately low LUMO to the designed products, favorable for inducing NIR optics. In these previous designs, benzene-1,2-diamine and 1,2,4,5-tetramine were applied to react with 4Br-NDI via their N-nucleophilic sites. Thus, polycyclic products were formed featuring both electron-accepting NDI and electron-rich dihydrophenazine moieties, bringing about the NIR-absorbing attributes.

In the current work, we integrate naphthalene-1,5-diamine with two equivalents of 4Br-NDI to construct a polycyclic NIR chromophore, via a tandem reaction entailing a nucleophilic aromatic substitution (S_{Ar}) followed by a unique metal-free C–C coupling. A large N-hetero-polycyclic tetramide molecule 1 (Scheme 1) exhibiting absorptions at ca. 1000 nm was thus obtained. Upon further functionalizing with 2-(dimercaptomethylene)malononitrile, molecule 2 was achieved, impressively manifesting major absorption around 1100 nm (ε > 10^5 mol^{-1} L cm^{-1}). Besides the...
very narrow optical bandgap, compound 2 also possessed an exceptionally low LUMO level at -4.72 eV. Such a low LUMO level greatly favoured the electron-transporting ability. A remarkable electron mobility up to 0.96 cm² V⁻¹ s⁻¹ was determined for 2 in a solution-processed thin-film transistor. The notable stability of 2 was underscored by comparison to molecule 3 with a similarly low LUMO level. Such a rare combination of high stability, strong NIR absorption, and optimal n-type semiconducting performance endows molecule 2 with great potential for opto-electronic applications.

Results and discussion

By subjecting naphthalene-1,5-diamine to excess 4Br-NDI, we were initially expecting a double SₐNAr product 1a’ (Scheme S1†), which was then planned to be transformed to 1 under separate Heck-coupling conditions. However, after working up the reaction of naphthalenediamine and 4Br-NDI in the presence of K₂CO₃, the H NMR spectrum indicated that two aromatic protons were missing from the expected 1a’. The mass spectroscopy also revealed that, while the generated molecule incorporated two NDI units, it contained only four bromine atoms. After analyzing the complete characterization results, we concluded that molecule 1 was produced in one pot from naphthalene-1,5-diamine and 4Br-NDI in the absence of the Pd catalyst (Scheme 1).

The mechanism of this tandem process was then examined in more detail. Molecule 4 was obtainable in decent yield at a shortened reaction time and lowered temperature, while 4a was not isolated. Since 4Br-NDI was known to undergo substitutions with various nucleophiles,¹⁰ SₐNAr between 4Br-NDI and naphthalenediamine reasonably happened (Scheme 2). Since debromination was not detected with 4a or 1a, the following intramolecular C-C coupling was unlikely to involve redox processes.¹¹ Moreover, it was found that molecule 1 was formed much faster in tetrahydrofuran than toluene, with nearly identical yields, which also suggested a polar mechanism. We thus proposed that the intramolecular C-C coupling was a SₐNAr process for the bromo-NDI moiety, with C(sp²)H as the nucleophile (Scheme 2). It was suspected that the strongly electron-donating NH was a critical activator in this electrophilic aromatic substitution of naphthalenediamine, by conferring high nucleophilicity to its para-position. To prove this hypothesis, we then carried out a reaction between 4Br-NDI and 1,3-phenylenediamine. As expected, molecule 5 was generated, substantiating the notion that NH activated its para-CH and realized a 5-membered ring annulation (Schemes 2 and S2†).
The UV/vis/NIR absorption spectrum of 1 showed a broad absorption band in the range of 600–1200 nm (Fig. 1a), and the maximum ($\lambda_{\text{max}}$) emerged at ca. 1000 nm, with a large molar extinction coefficient ($\epsilon$) of $1.2 \times 10^5$ L mol$^{-1}$ cm$^{-1}$. The vibronic structures were clearly observable, consistent with the highly rigid polycyclic skeleton of the chromophore.$^{5,12,13}$

Impressed by the remarkable NIR optical properties of 1, we were intrigued to investigate whether the bandgap could be further narrowed through chemical modifications. After various attempts, molecule 2 was successfully prepared by subjecting compound 1 to sodium 1,1-dicyanoethylene-2,2-dithiolate. Such a modification was anticipated to further lower the bandgap since the resultant 2 possessed a further expanded polycyclic $\pi$-system with more intense D-A characteristics. Desirable results were observed when the absorption spectrum of 2 was collected. While the band shape remained quite similar to that of 1, the overall spectrum was shifted to a longer wavelength by about 100 nm, giving rise to $\lambda_{\text{max}}$ at ca. 1100 nm with $\epsilon > 1.3 \times 10^5$ L mol$^{-1}$ cm$^{-1}$ (Fig. 1a). Such strong absorptions around 1100 nm are observed for the first time with closed-shell polycyclic dicarboximide dye molecules.

The absorption spectra of 1 and 2 at varied concentrations suggested that these molecules were weakly aggregating in chloroform (Fig. S1 and S2†). Both 1 and 2 were weakly luminescent, exhibiting small Stokes shifts of 412 and 257 cm$^{-1}$, respectively (Fig. S4 and S5†).

Subsequent electrochemical study unveiled that the much narrowed bandgap of 2 was mainly attributable to its substantially lowered LUMO level. As shown by the cyclic voltammograms (CV), both 1 and 2 displayed two reduction and two oxidation waves (Fig. 1b and S8†). All these redox processes were reversible. A particularly low LUMO at $-4.72$ eV was displayed by 2, in comparison to the LUMO at $-4.35$ eV for 1 (Table 1). On the other hand, the HOMO energy levels of the two molecules were separated by $0.03$ eV, which explained the much narrower bandgap of 2. It is noteworthy that, in spite of such a low-lying LUMO and narrow bandgap, compound 2 was fairly stable under ambient conditions. No detectable changes were observed in the absorption or NMR spectra after storing under ambient conditions for months.

The remarkable stability of 2 was further underlined by comparison to molecule 3 with a similarly low LUMO. Compound 1 could be oxidized to 3 in nearly quantitative yield using PbO$_2$ (Scheme 1). This redox process was well reversible, and the reduction of 3 back to 1 was realized, also quantitatively, with 1,4-phenylenediamine. The absorption spectrum showed that molecule 3 also possessed a low-energy $S_0 \rightarrow S_1$ band in the NIR regime, exhibiting a local maximum at about 1000 nm, but this low-energy transition displayed minimal extinction ability (Fig. 1a). The overall absorption maximum at a much higher energy emerged around 525 nm. Not surprisingly, CV revealed that the dehydrogenated molecule 3 displayed a considerably lowered LUMO at $-4.66$ eV compared to 1, actually slightly higher than that of 2 (Table 1). However, unlike compound 2, inferior chemo-stability was observed with 3. Both absorption and NMR spectra indicated that molecule 3 was partially reduced to 1 after storing under ambient conditions for only a few days. The similar LUMO level but superior

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**Fig. 1**  (a) UV-vis-NIR absorption spectra (at 1.0 × 10$^{-5}$ M in CHCl$_3$); (b) cyclic voltammograms of 1–3 recorded in CHCl$_3$.

**Table 1**  Optical and electronic properties

|   | $\lambda_{\text{abs}}$ [nm] | $\epsilon$ [mol$^{-1}$ L cm$^{-1}$] | $\lambda_{\text{em}}$ [nm] | LUMO$^d$ [eV] | HOMO [eV] | LUMO–HOMO$^f$ [eV] | $E^g_{\text{f}}$ [eV] |
|---|---|---|---|---|---|---|---|
| 1 | 1000 | $1.2 \times 10^5$ | 1043 | $-4.35$ | $-5.27$ | 0.92 | 1.09 |
| 2 | 1101 | $1.4 \times 10^5$ | 1133 | $-4.72$ | $-5.30$ | 0.58 | 0.99 |
| 3 | 1004 | $4.7 \times 10^4$ | — | $-4.66$ | $-5.80$ | — | 1.14 |

$^a$ Absorption maxima of the $S_0 \rightarrow S_1$ transition in CHCl$_3$ solution. $^b$ Molar extinction coefficient at $\lambda_{\text{abs}}$. $^c$ Emission maxima in CHCl$_3$. $^d$ Data from CV. $^e$ Optical bandgap from the absorption onset. $^f$ Calculated from the optical bandgap and LUMO in CV.
chemo-stability of 2 compared to 3, by virtue of the dihydro-
structure, further stressed the precious values of the former.

Time-dependent density functional theory (TD-DFT) calcula-
tion results confirmed the experimental observations by
showing that the $S_0 \rightarrow S_1$ transition in compound 3,
corresponding to electron excitation from HOMO to LUMO,
possessed a much smaller oscillator strength compared to its
higher energy transitions (Fig. S14†). DFT and TD-DFT calcula-
tions were also performed for 1 and 2. Remarkably, the HOMO
and LUMO of 1 and 2 were extensively delocalized over the
total polycyclic π-framework (Fig. 2). Consistent with the
experimentally observed strong NIR absorbing abilities of 1
and 2, TD-DFT calculations also verified that both molecules
manifested very large transition dipole moments and significant
oscillator strengths with their low-energy $S_0 \rightarrow S_1$ (HOMO to
LUMO) transitions.

The low-lying LUMO level and delocalized frontier orbitals" of
2 prompted us to examine its electron-transporting semi-
conducting capability. To this end, organic field-effect tran-
sistors (OFET) with top-gate/bottom-contact configuration were
fabricated. Using the solution processing technique, the active
layer was deposited by spin-casting solutions of 2 in trichloro-
ethylene (10 mg mL$^{-1}$) on patterned Au(source–drain)/SiO$_2$/Si
substrates. After thermal annealing the semiconducting mate-
rials, a poly(perfluorobutenylylene) (CYTOP) solution was
spin-coated on top of it as the dielectric layer, followed by
thermally evaporating a layer of aluminum as the gate electrode.
All devices were fabricated in a glove box but tested under
ambient conditions ($R_{H} = 50–60\%$). As expected, molecule 2
exhibited typical n-type transport characteristics. Quite
impressively, optimal electron mobility ($\mu_e$) up to 0.96 cm$^2$ V$^{-1}$
s$^{-1}$ (Fig. 3) and an average $\mu_e$ of 0.93 cm$^2$ V$^{-1}$ s$^{-1}$ were deter-
mined. In comparison, an electron mobility of merely 0.007 cm$^2$
V$^{-1}$ s$^{-1}$ (Fig. S11†) was measured for 1 under similar conditions,
while compound 3 was not completely stable (partially reduced)
during the device fabrication.

Notably, the transfer and output characteristics of 2 showed
negligible hysteresis, which has rarely been observed for n-type
organic materials and could be attributable to the low LUMO of
the molecule. Besides, no contact resistance was observed in the
output curves, suggesting good contact between molecule 2 and
the gold electrode.

In the device characterizations, it was noticed that the elec-
tron mobility of 2 was highly sensitive to the annealing
temperature (Fig. S10†). TGA and DSC characterizations
confirmed adequate thermal stability of molecule 2 (Fig. S9†),
so a relatively wide range of annealing temperatures was
examined. When the semiconductor was annealed at 100 °C,
only a moderate electron mobility of 0.12 cm$^2$ V$^{-1}$ s$^{-1}$ was ob-
tained. Whereas, if the annealing temperature was elevated to
150 °C, the electron mobility was significantly improved to 0.73
cm$^2$ V$^{-1}$ s$^{-1}$. More desirable performances were achieved
between 180 and 250 °C, with the mobility fluctuating in the
range of 0.8–1.0 cm$^2$ V$^{-1}$ s$^{-1}$ (Fig. S10†). Moreover, at the opti-
ized annealing temperature, very low $V_{th}$ values of $-5$ to $0$ V
were observed. Subsequently, AFM and X-ray diffraction studies
elicited that the variation in the device performance was
Clearly correlated to the crystallinity of the semiconducting

![Fig. 2](image-url)  
**Fig. 2** DFT calculated geometry (side view) and HOMO/LUMO (top view) of 1 and 2 (alkyl side groups are replaced by methyl in the calculations).

![Fig. 3](image-url)  
**Fig. 3** (a) Transfer ($V_{DS} = 100$ V) and (b) output profiles of 2 (annealed at 220 °C) in OFET ($\mu_e = 0.96$ cm$^2$ V$^{-1}$ s$^{-1}$).
layer. Highly crystalline morphology and best device performance were both obtained after annealing at about 220 °C (Fig. 4 and 5). Such high electron mobility, as well as the low V_{TH} value, is believed to benefit from the low-lying LUMO and suitable frontier orbital distribution of 2. The S⋯S interactions may have helped to induce a favorable molecular packing motif.

Conclusions

In conclusion, a neutral organic small molecule 2 exhibiting strong NIR absorption around 1100 nm (ε = 10^5 mol⁻¹ L cm⁻¹) and an electron mobility up to 0.96 cm² V⁻¹ s⁻¹ is developed. The synthesis of the molecule is accomplished via a tandem process, involving a unique metal-free C–C coupling reaction between an electrophilic aryl bromide and an electron-rich aryl amine with dual nucleophilic sites of NH and CH groups. Having an exceptionally low-lying LUMO at ~4.72 eV, the advantageously high chemo-stability of 2 is highlighted by comparison to molecule 3, which has a slightly higher LUMO than 2 but undergoes auto-reduction under ambient conditions. Such a combination of low-energy NIR-absorption, n-type semiconducting ability and optimal chemo-stability is rarely available for organic molecules. To the best of our knowledge, this is the first example of an organic small molecule that manifests both high performance n-type semiconducting properties and strong NIR absorption at wavelengths exceeding 1 μm. Such distinctive properties qualify the molecule for special applications as a transparent organic opto-electronic material.

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