Review Article

Recent Progress of Graphene-Based Photoelectrode Materials for Dye-Sensitized Solar Cells

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Graphite with a single atomic layer known as graphene shows great capability in energy conversion and storage devices. Dye-sensitized solar cells (DSSCs) have attracted intense interests due to offering high photo-to-electric conversion efficiencies. DSSCs are built from a photoelectrode (a dye-sensitized nanocrystalline semiconductor), an electrolyte with redox couples, and a counterelectrode. In this review article, we outline the strategies to enhance the efficiency and reduce the cost by introducing graphene into the DSSCs as the photoelectrode. First, the development of DSSCs and the properties of graphene are briefly described. Then, the applications of graphene-based materials for photoelectrodes (transparent electrode, semiconductor layer, and dye sensitizer) in DSSCs are deeply discussed. Finally, an outlook for graphene materials in DSSCs is provided.

1. Introduction

Finding an alternative option of energy has become today’s essential need, with depleting conventional resources of energy, such as fossil fuels. Solar energy, as the largest single available source of clean energy, has the potential to overcome this problem. Solar cells provide an efficient method to convert solar energy into electricity [1]. The first-generation solar cells are made of ultrapure silicon metal. However, the production of ultrapure Si requires high energy, leading to the cost of electricity produced by this technology which is several times higher than that produced from fossil fuels [2]. One of the alternatives for the Si solar cells is dye-sensitized solar cell (DSSC), because of their relatively high photo-to-electric conversion efficiencies, low production cost, and environmental benignity [3, 4]. DSSC basically consists of three components, including photoelectrode, electrolyte, and counterelectrode. O’Regan and Grätzel established the state-of-the-art DSSCs: a dye-sensitized nanocrystalline TiO2 film as photoelectrode, I3-/I- redox couple as electrolyte, and Pt as counterelectrode [5, 6]. Applications of new materials have been explored to enhance the efficiency and reduce the basic cost of DSSCs.

1.1. Dye-Sensitized Solar Cells (DSSCs). DSSCs are also known as Grätzel solar cells, honoring their invention and the pioneering work. Figure 1 shows a schematic representation of a DSSC, including a photoelectrode, a photosensitizer, an electrolyte, and a counterelectrode [7]. A DSSC has three important steps through which it converts sunlight into electrical energy: when sunlight falls on a dye, it starts the photoexcitations and makes electrons to move to the semiconductor’s conduction band. Dye molecules are oxidized back by electrons given by electrolyte through redox reaction and finally travel through the external load electrons to complete the circuit. A number of reviews and book chapters on DSSCs have been published in the literature [8–11]. The electrochemistry, physics, materials science, and technology behind DSSCs have been extensively reviewed therein by leading professionals working in this field. In this review article, we are going to review the latest developments of graphene photoelectrode materials.

1.1.1. The Function of Photoelectrode and Its Requirements. The photoelectrode is a mesoporous oxide layer composed of nanometer-sized particles with a monolayer of dye...
attached to the surface, which is responsible for light absorption. It is an important component in DSSC, as it converts photons into electrical energy [12, 13]. It can strongly influence the photovoltage, the fill factor, and the photon-to-current conversion efficiency (IPCE). The process can be described as follows: when exposed to sunlight, the electrons of the dye are first excited from the highest occupied molecular orbital (HOMO) to the lowest unoccupied molecular orbital (LUMO) [14]. Then they are transferred to the semiconductor’s conduction band and passed from the semiconductor network, being accumulated at the transparent conducting oxide. An electrolyte solution which typically consists of an iodide/ triiodide redox couple donates electrons to regenerate the oxidized dye. At last, the electrons transport to the counter electrode through the external circuit and reduce the triiodide ions back to the iodide to complete the process. A good photoelectrode is supposed to provide good electron injection, good electron collection, and good usage of light. Many different materials have been explored as photoelectrode in DSSCs. Among them, titanium dioxide (TiO₂) powder is most abundantly used because of the low recombination rate for the hole-electron pair and great absorption property [15]. Moreover, ZnO which has a very high electron mobility has also gathered much attention to be employed as photoelectrode materials in DSSCs [16–36].

1.2. Graphene. Graphene, one of the allotropes of abundantly available carbon, has emerged as one of the most promising materials for applications in solar cells. It was first discovered in 2004 by Novoselov et al. [37] and awarded with a Nobel prize in 2010 [38–40]. Graphene is a 1-atom-thick transparent layer of sp²-hybridized carbon atoms packed into a 2D nanostructure (Figure 2) [39]. The related structures include fullerene, carbon nanotubes, and graphite (Figure 2). Graphene possesses high carrier mobility, low sheet resistance, and high optical transparency, which would constitute its excellent candidate as an electrode material [37]. Another unique property graphene has is that it can repair holes in the sheet by itself when it is exposed to the molecules containing carbon atoms [41]. Carbon atoms align themselves in a hexagonal lattice perfectly. Pristine graphene has no bandgap; therefore, it acts as a semimetal [42]. Currently available methods aiming for pristine graphene include mechanical exfoliation [43] and liquid exfoliation. However, both of them have low production. Scientists are exploring new chemical and physical ways to create an artificial bandgap in graphene, which is one of the requirements for the fabrication of electronic devices. Zero-bandgap graphene can be transformed into a wide-bandgap semiconductor through hydrogenation via sp³-C–H bond formation [44].

Because graphene is an atom-thick layer [45], it is a perfect nanoscale material and, therefore, has great potential in a very wide range of applications in the field of nanotechnology, including nanoelectronics, nano optics, display devices, LEDs, computer data storage, energy, membranes, nanofilters for water purification, sensors, nanomedicine, stem cells, and energy conversion devices [46–50], such as DSSCs [51–56]. Graphene possesses a lamellar structure and high conductivity, which can help to improve the conductivity and increase the photovoltaic efficiency of DSSCs (5.98%) when introduced to MoS₂ [57]. Graphite paper is another form of graphene. It is very lightweight and has a flexible structure which is good for mass production and for use in DSSCs [58]. Reduced graphene oxide (rGO) is an intermediate stage of graphene oxide and graphene, and it has various types of oxygen-containing functional groups, like −OH, =O, and −COOH, and lattice surface defects which generate the electrocatalytic site in metal nanoparticles. This is why rGO showed much better performance than the fully reduced defect-free graphene [59]. Graphene nanoplatelets (GNP) have been explored as counter electrodes because of their higher active sites, such as edge defects and oxide groups [60].

1.2.1. Preparation of 2D Graphene. In order to reach a high mass production capability and understand the process which involves a solution or thin film basis, numerous efforts have been developed to synthesize graphene and its derivatives. These processes are usually differentiated in two categories: the bottom-up and top-down approaches. Graphene materials can be directly synthesized from the carbon sources
in a bottom-up approach. For example, chemical vapor deposition (CVD) is a generally bottom-up process used to develop large-area, single, and few-layer graphene sheets on metal substrates [61–64]. Another example is the plasma-enhanced CVD (PECVD). The difference between PECVD and CVD is that this method can develop one-layer graphene in large quantity at a lower deposition temperature and in a shorter reaction time. Besides, single-layer and few-layer graphene films can also be obtained via the graphitization of carbon-containing substrates, such as SiC, through high-temperature annealing [65, 66]. In addition to these solid-phase deposition methods, wet chemical reaction of ethanol and sodium followed by pyrolysis is another method to synthesis graphene [67], or graphene-like polyaromatic hydrocarbons through organic synthesis [68, 69]. Figure 3(a) shows the electrical characteristics of the as-prepared transparent carbon films via a new bottom-up chemical approach involving the thermal reaction of synthetic nanographene molecules of giant polycyclic aromatic hydrocarbons which are cross-linked with each other and further fused into larger graphene sheets (Figure 3(b)). A small decrease in the film conductivity was observed as the film thickness decreased. The structure of the graphene film, as shown in Figure 3(c), consisted of ordered, tightly packed graphene layers. As illustrated in Figures 3(d) and 3(e), graphene film shows an ultrasmooth surface compared with the relatively rough surface and chemical instability of ITO [68].

Compared with the bottom-up approaches, the top-down approaches show great potentials, such as high yield, liquid-based processability, and less complicated execution. Intercalation, chemical functionalization, sonication of bulk graphite, and electrochemical exfoliation of graphene have been demonstrated as top-down approaches [70]. However, there are some drawbacks of top-down methods, such as low yield of single-layer production [71], expensive intercalates [72], and residual surfactant-induced low conductivity.

Therefore, the reduction of highly oxidized graphene oxide (GO) sheets from the exfoliated graphite oxide has been developed as an alternative approach [73]. The reaction of graphite with a mixture of potassium permanganate (KMnO₄) and concentrated sulfuric acid (H₂SO₄) is usually referred to as Hummers’ method. Graphite oxide can be obtained by this method [74, 75]. The surfaces of exfoliated GO sheets are thus highly oxidized and featured with the residual epoxides, hydroxides, and carboxylic acid groups [76, 77]. Till date, rGO sheets can be obtained by various types of reduction methods, such as chemical reagent reduction [78–86], photochemical reduction [87–90], microwave-assisted reduction [91], thermal reduction [92–98], photothermal reduction [99, 100], and electrochemical reduction [101–103]. Many reduction agents were employed in the chemical reduction process, such as hydrazine [104], strong alkaline media [83], vitamin C or ascorbic acid [105], bovine serum albumin (BSA) [85], bacterial respiration [106], and hydriodic acid [107]. Hydriodic acid is used to change the hydrazine reduction from the reaction of making rGO thin films due to its low toxicity. Although rGO has more charge-transfer resistance than pure graphene because of defects and the presence of residual oxygenated groups, the tenability in electronic and optoelectronic properties via chemical reactions [108] and the feasibility for composite incorporation [109] can be provided by the reactive surfaces.

1.2.2. Preparation of 3D Graphene. There are various requirements according to different applications to satisfy them; three-dimensional (3D) graphene with various morphologies, structures, and properties was designed and synthesized via numerous methods.

There are various strategies used, but self-assembly is one of the most common for bottom-up nanotechnology. Graphene naturally becomes a versatile nanoscale building block for self-assembly to achieve hierarchical microstructures and novel functionalities due to its unique structure and properties [110]. One very common example for producing 3D graphene structures is through the gelation process of GO dispersion followed by reduction to convert GO to rGO [111]. The lyophilized GO dispersion shows a 3D network composed of GO sheets, as shown by its SEM images (Figure 4). GO sheets can be nicely distributed in water-based solvent because of the force balance between the van der Waals attractions from the basal planes of GO sheets and the electrostatic repulsions from the functional groups of GO sheets. Once this force balance is broken, gelation of GO dispersion occurs. During the gelation process to form a 3D design of GO hydrogels, GO sheets overlap each other a little bit. 3D rGO networks are obtained after further reduction of the GO hydrogels. Many methods have been employed to initiate the gelation of GO dispersion, for example, addition of cross-linkers [112], change in the pH value of the GO dispersion [112], or ultrasonication of the GO dispersion [113]. In the meantime, so many other materials have also been explored as cross-linkers for the self-assembly of GO sheets, such as DNA [114], metal ions [115–117], polymers [118], and organic molecules [119]. In addition to the gelation process of a GO dispersion, other methods such as...
freeze-drying [120], tape casting [121], controlled filtration [122], centrifugation of GO dispersions [123], electrochemical deposition [124], and sol-gel reaction [125, 126] have also been employed to obtain 3D graphene materials.

Another method that can prepare 3D graphene is called the template-assisted method. The advantage of this method is that it can obtain 3D graphene with much more controlled morphologies and properties because of the use of pre-designed 3D templates, such as CVD methods [127–129]. For example, 3D graphene networks were successfully synthesized by using commercially available Ni foam as both template and catalyst [127]. The synthesis of graphene foam and its integration with polymers are illustrated in Figure 5. The authors chose nickel foam, a porous structure with an interconnected 3D scaffold of nickel, as a template for the growth of graphene foam. Then, carbon was introduced into a nickel foam by decomposing CH$_4$ at 1000°C under ambient pressure, and graphene films were then precipitated on the surface of the nickel foam. Besides the Ni foam mentioned before, anodic aluminum oxide [130], MgO [131], nickel-coated pyrolyzed photoresist film [132, 133], metal nanostructures [134, 135], and metallic salts [136] have also been used as templates to synthesize 3D graphene.

In addition, the direct deposition of 3D graphene architectures by plasma-enhanced CVD (PECVD) methods [137–139] is a straightforward strategy, in which Au and stainless steel have been employed. 3D graphene was formed since the graphene sheets were vertically grown on the substrate and connected with each other. As a result, the materials obtained by this method firmly adhered to the substrate.

2. Graphene Photoelectrode Applications

The photoelectrode of a DSSC is usually made from transparent conducting glass or plastic substrate on which a dye-sensitized TiO$_2$ layer is coated. It is important to increase the loading of dye molecules, enlarge the interface area of the dye/electrolyte, and improve the conductivity of electrons at the semiconductor layer to compete with charge recombination. Graphene possesses excellent properties like high transparency, good thermal stability, high room temperature carrier mobility, and low impedance level, which have attracted great attention [140]. Recently, in photodelectodes graphene material also has been explored as transparent conductors, as electron promoter in the semiconductor layer, and as additive in dye sensitizers.

2.1. Transparent Electrode. Indium tin oxide (ITO) material has high conductivity and very high transmittance in the visible spectrum, and that is why it is used as transparent electrode much widely. But the problem with the ITO material is that it is brittle and unstable at high temperature. So that
Figure 4: SEM images of lyophilized GO solution and three typical GO hydrogels: (a) GO solution, (b) GO/PVP hydrogel with 1 mg mL$^{-1}$ PVP, (c) GO/PDDA hydrogel with 0.1 mg mL$^{-1}$ PDDA, and (d) GO/Ca$^{2+}$ hydrogel with 9 mM Ca$^{2+}$. $C_{GO} = 5$ mg mL$^{-1}$, and scale bar = 10 μm [111].

Figure 5: Synthesis of a GF and integration with PDMS. (a) A nickel (Ni) foam. (b) CVD growth of graphene films (Ni–G) using Ni foam as a 3D scaffold template. (c) An as-grown graphene film after coating a thin PMMA supporting layer (Ni–G–PMMA). (d) A GF coated with PMMA (GF–PMMA) after etching the nickel foam with hot HCl (or FeCl$_3$/HCl) solution. (e) A free-standing GF after dissolving the PMMA layer with acetone. (f) A GF/PDMS composite after infiltration of PDMS into a GF. All the scale bars are 500 μm [127].
is why graphene has been coming out as an alternative material which has a low cost and has better properties than the ITO material without any limitations [141].

Wang et al. were the first ones to make a transparent and conductive graphene electrode in 2008 for DSSCs [142]. The graphene films were fabricated via the exfoliation of graphite oxide and thermal reduction of the resultant platelets. As the transparent electrode, solid-state DSSCs based on spiro-OMeTAD and porous TiO₂ were fabricated as shown in Figure 6. The work function of graphene is lower than that of the counter gold electrode, therefore qualifying for application as the transparent anode in the DSSCs as indicated in Figure 6(a). An efficiency of 0.26% was obtained when a graphene film with a Rs of 1.8 kΩ/sq and transmittance of 72% (550 nm) was used as anode electrode. The low efficiency of DSSC was claimed to be due to the low quality of the graphene film [142]. Furthermore, thin-layer graphene films on SiO₂ substrates were prepared by Huang et al. by using ambient pressure chemical vapor deposition. The prepared graphene films achieved a power conversion efficiency of 4.25% which is not so much different from those FTO counter-electrodes [143].

2.2. Semiconducting Layer. To increase the performance of the DSSC nowadays, graphene has been added to the semiconductor which improves the charge collection [144–146]. The work function of graphene, which is about -4.4 eV, is between the conduction band of TiO₂ and the work function of ITO. This benefits in suppressing the charge recombination by fast collection of the electrons at the anode [147]. On the other hand, visible light irradiation can excite the valence electrons from graphene into the TiO₂ conductive band at the graphene/TiO₂ interface based on a theoretical study [148], which results in separated electron-hole pairs. In addition, the conductive percolation threshold of the graphene/TiO₂ layer is only at 1 vol% of the graphene loading [149]. However, transmittance of the composite film decreases with increasing the graphene content; hence, to have a maximum efficiency, the optimal value needs to be obtained [149, 150]. Besides the enhanced conductivity, the large surface area of graphene improves the loading and dispersion of the dye molecules. Graphene processes a theoretical surface area of 2630 m² g⁻¹, which shows great potential as an ideal support material with enhanced interfacial contact even when used in small amounts [151]. It is reported that graphene surfaces can bind with dye molecules, such as porphyrin. Then, the photocurrent will be generated when photoirradiation undergoes energy and electron transfer [152]. Moreover, the light scattering at the photoanode can also be enhanced by forming a graphene/TiO₂ composite porous network [149]. Consequently, by using graphene/TiO₂ as the photoanode, efficiency was increased by 39% higher than using commercial P25 TiO₂ which is 4–7% [149].

There are so many ways through which graphene has been incorporated into the semiconductor film. Kim et al. reported the UV-assisted photocatalytic reduction of graphene oxide/TiO₂ nanoparticles. In the gap of FTO and nanocrystalline TiO₂, the prepared graphene/TiO₂ was applied as an interfacial layer [153]. Because of the lower roughness of graphene-TiO₂, it provided much better adhesion than did the normal interfacial layer between FTO and TiO₂. The solar cell showed a PCE of 5.26%, which is higher than that without a blocking layer (PCE = 4.89%). Chen et al. also used a similar kind of method to increase the performance of the DSSCs. They spin-coated the graphene on the FTO which helps in preventing the recombination of the charge between FTO and TiO₂ [144]. Because of this, efficiency improved marginally from 5.80% to 8.13%.

Graphene could also be used as a current collector. Yang et al. incorporated 2D graphene into the TiO₂ nanostructure, aiming at increasing light collection, preventing charge recombination, and thus enhancing the charge transportation rate [154]. Bavir and Fattah employed a composite of TiO₂ and graphene as photoelectrode to replace the TiO₂-ZnO composite [18]. DSSC with TiO₂-graphene as photoelectrode shows a higher efficiency of 9.3% than that (6.5%) of TiO₂-ZnO photoelectrode. Table 1 summarizes the photovoltaic characteristics of DSSCs fabricated with graphene incorporated into the semiconducting layers.

Low et al. used rGO and TiO₂ nanocomposite as photoanode [155]. They tried to find out the optimum amount of TiO₂ on the rGO photoanode for maximum efficiency. 0.3 wt% of TiO₂ on rGO showed a maximum photocurrent generation efficiency. rGO has limited capacity to absorb visible light from solar illumination, and it is only 2.3%. Adding TiO₂ enhances the capacity of rGO to absorb the light. Here, graphene oxide was synthesized through improved Hammer’s method from graphite flakes. In the acids H₂SO₄ :H₃PO₄ (9 : 1), a mixture of graphite flakes (1.5 g) and KMnO₄ (9.0 g) was poured, and the reaction was carried out in cold water. The whole reaction was carried out in normal cold water (<20°C) and stirred for 24 hours. By adding 3 mL of H₂O₂ in the mixture, the oxidation process was stopped. The color of the mixture became light brownish showing a high oxidation level of graphite. GO was centrifuged to remove the supernatant. HCl and DI, remaining solid, were washed and dried in the oven for 36 hours. The dried powder is then again added with water (3 mg/mL). Then, hydrazine (1 μL for 3 mg of GO) is added to the mixture, and to add more heat, it was immersed to an oil bath, with oil at 80°C. The obtained mixture in the last was high-quality RGO. For the preparation of TiO₂, TTIP was used as a precursor and the precipitation-peptization method was used. The mixture was made by adding 10 mL of TTIP and 40 mL of propanol solution then stirred for 30 minutes. Another mixture was made by adding 5 mL of acetic acid, 10 mL of propanol, and one drop of Triton X-100 into the purified water. This mixture was also stirred for 30 minutes. The first mixture was then added slowly into the second mixture with a rate of 1 mL/min. Finally, TiO₂ solution was obtained by stirring the mixture for 2 hours. Different samples were made by adding various amounts of TiO₂ (0.1, 0.2, 0.3, 0.4, and 0.5 wt% of TiO₂, respectively) solution into the RGO solution and dried into the oven for 24 hours. Samples were then calculated at 450°C for 2 hours. Figure 7(a) shows the complete work of the DSSC schematically under a light irradiation process. At the beginning, the adsorption of a photon by N-719 dye will be transformed from the dye
molecule, S, to the excited state (S^*) during the harvesting of DSSCs under the irradiation with AM 1.5 condition. Then, an electron was transferred rapidly (ps or fs) to the wide bandgap semiconductor. After that, from the N-719 dye an excited electron, S^*, was injected into the TiO_2–rGO photoanode while the other S^* collected by FTO glass followed through the outer circuit heading to the cathode (graphite) of the FTO glass. Finally, electrons through electrolyte redox reaction I/I_3 reduce the oxidized S^* to the original state. Performance of DSSC with 0.3% wt of TiO_2 and RGO photoanode has a maximum efficiency of 7.2%, with J_{sc} of 28.36 mA cm^{-2}, V_{oc} of 0.54, and ff of 0.47 (Figure 7(b)). This is mostly because of the higher probability to capture and transport the electrons from photoelectrode in a high-conductivity condition. The conductivity was maximized because of the formation between rGO and TiO_2; hence, rGO shows a very high conductivity and TiO_2 has high photocatalytic characteristic.

Tang et al. proposed a molecular grafting method to develop a photoanode where exfoliated graphene sheets (GS) are attached to a TiO_2 nanoparticle matrix in which the graphene sheets were chemically exfoliated and chemisorbed in

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**Figure 6:** Illustration and performance of a solar cell based on graphene electrodes. (a) Illustration of the dye-sensitized solar cell using graphene film as electrode; the four layers from bottom to top are Au, dye-sensitized heterojunction, compact TiO_2, and graphene film. (b) The energy level diagram of the graphene/TiO_2/dye/spiro-OMeTAD/Au device. (c) I-V curve of the graphene-based cell (black) and the FTO-based cell (red), illuminated under AM solar light (1 sun) [142].

**Table 1:** PV characteristics of DSSCs fabricated with graphene incorporated with a semiconducting layer.

| Photoelectrodes | J_{sc} (mA cm^{-2}) | V_{oc} (V) | ff | η (%) | Ref. |
|-----------------|---------------------|------------|----|-------|------|
| Underlayer with RGO, scaffold layer with RGO, scattering layer with RGO | 23.2 | 0.73 | 0.55 | 9.2 | [163] |
| Scaffold layer with T-CRGO | 8.4 | 0.75 | 0.68 | 4.3 | [156] |
| Scaffold layer with CVD-derived graphene on Al_2O_3 | 10.2 | 0.78 | 0.68 | 5.4 | [164] |
| Scaffold layer with TRGO | 7.6 | 0.67 | 0.54 | 2.8 | [146] |
| Scaffold with thermally treated CTAB-functionalized graphene | 12.8 | 0.82 | 0.62 | 6.5 | [165] |
| Scaffold layer with T-CRGO | 16.3 | 0.69 | 0.62 | 7.0 | [150] |
| Underlayer with T-CRGO | 6.7 | 0.56 | 0.4 | 1.7 | [147] |
the TiO$_2$ matrix (Figure 8) [147]. The conductivity of the reduced graphene sheets and good attachment of TiO$_2$ could be achieved by controlling the oxidation time of the chemical exfoliation process. It was proved that by adding the GO not only the efficient electronic transport path was provided but also the increased dye loading of the film, leading to an increase in the photocurrent. The graphene-implanted TiO$_2$ showed a conversion efficiency of 1.68%. Graphene was also reported to be dispersed using Na and then incorporated in TiO$_2$ particles by heterogeneous coagulation particles [156]. It is known that P25 particles and graphene process opposite zeta potentials; therefore, the TiO$_2$ nanoparticle binds to the surface of graphene via a strong electrostatic attractive force. The DSSC with P25 nanoparticles showed a $J_{sc}$ of 5.04 mA cm$^{-2}$ and PCE of 2.7%. In the presence of 0.5 wt% graphene, $J_{sc}$ increased from 66% to 8.38 mA cm$^{-2}$, resulting in an efficiency of 4.28%. The authors proposed that the creation of surface morphologies increased the available sites, while electrons travel through long mean free paths without a recombination-extended electron lifetime. As a result, the performance was enhanced. Figure 9(a) shows the TEM images of graphene, which are thin, and part of them overlay while containing wrinkles and rolled edges. Figure 9(b) shows the zeta potential curves for the graphene and P25 particles in ethanol. The zeta potential of graphene is -42 mV, while that of P25 is 15 mV. Clearly, because of strong opposite potentials on both the materials, particles of P25 tend to bind on the surface of graphene. Figure 9(c) shows EIS spectra of the DSSCs using P25 and P25–graphene electrodes at an applied bias of $V_{oc}$. In a high-frequency region and middle-frequency region, there are two well-defined semicircles. Acceleration of the electron transfer process in the film photoanode can be seen from the reduced size of the semicircle in the high-frequency region for the P25–graphene electrode compared to the P25 electrode. As shown in Figure 9(d), The work function of graphene is in the range of 4.42-4.5. It signifies possible electrons moving from the conduction band of P25 to graphene. Then, the electrons transfer through the graphene sheets and were collected by FTO glasses.

Kilic and Turkdogan compared the results of DSSC with FTO/3D-ZnO-based photoanode and graphene/3D-ZnO--based photoanode [157]. Adding graphene increases the porosity of the photoanode which helps in increasing the charge transfer. Graphene was synthesized from the graphite by first converting it into reduced graphene oxide and then into graphene flakes by the modified Hummers’ method. This graphene was then dip-coated on the FTO substrate. Then, by facile hydrothermal growth method, 3D-ZnO was grown on graphene. Efficiency of DSSC with graphene/3D-ZnO drastically increased than using pure ZnO as photoanode, which was improved from 5.117% to 6.628%. Because of the good interconnectivity and fast charge transport between graphene and the 3D-ZnO nanostructure, there is an increase by 29.4% in PCE for graphene/ZnO photoanode DSSC compared to DSSC with conventional ZnO electrode and Pt CE. It was proved that the graphene layer increased the external quantum efficiency with more charge carrier collection through a better aligned band structure, which reduces the charge recombination rate.

2.3. Dye Sensitizer. There are two requirements for an effective sensitizer of a DSSC: (1) it should have the capability to absorb a wide range of solar spectrum, and (2) electrons must be quickly moved to the semiconductor scaffold from the ground state. Traditionally, ruthenium polypyridyl and quantum dots are used as sensitizers in the photoanode for
DSSC. It was reported that graphene has the ability to work as photosensitizer because charge injection from the graphene material to TiO₂ can occur. However, the performance of graphene in this role is poor. Nevertheless, they have recently shown promise in hot injection [158], showing the promising ability to overtake the Shockley-Queisser efficiency limit inherent to current device structures.

Yan et al. fabricated DSSC using graphene material photosensitizers for the first time [159]. The graphene quantum dots (GQDs) which have a diameter ranging from 1 to 30 nm and had an absorption maximum at 591 nm were employed in this device. The extinction coefficient is 10⁵ M⁻¹ cm⁻¹, which is almost an order of magnitude higher than N719. Figure 10 shows the employed strategy to make large graphene quantum dots soluble. In addition to that, the HOMO level was found lesser than the iodide/triiodide redox potential and the LUMO level was higher than the TiO₂ conduction band which suggested that both injection of electron and dye regeneration are possible. It was also proved by the same group that controlling the size of the molecule can tune the bandgap of GQD and changing its functionalization can tune the redox potential [160].

Zamiri and Bagheri also tried to use the combined effect of graphene material and quantum-sized properties for increasing the DSSC efficiency [161]. GQD has a few-graphene-layer size of 100 nm. They are the combination of unique properties of quantum-sized dots and graphene. At the edge, GQDs have very good functional groups, because of which they have an amazing water solubility. By putting multiple coatings of the ZnO nanoparticles on the FTO glasses and also then immersing it in N-719 and GQDs, the effect of the thickness of ZnO nanoparticles on the performance of the DSSC was studied. The I-V tests confirmed that photoanodes with 40 μm thickness show the highest efficiency because with increasing thickness, more photosensitizer molecules present in the semiconductor layer for absorbing sunlight. To study the effect of the various photosensitizers on ZnO photoanode, it was immersed in different photosensitizers for different times. The I-V test shows that if the immersion time is longer, more photosensitizers will be absorbed, which is helpful in increasing the light harvesting efficiency, which ultimately results in increasing efficiency of DSSC.

Esmaeili and Gaznagi tried to add graphene in those traditional sensitizers to improve efficiency [162]. They made two samples for the experiment. In the first sample, they just added graphene into the TiO₂ blocking layer, and in the second sample, they added graphene in the ruthenium-based dye solution as well. The authors demonstrated that loading graphene within the lonely blocking layer and the TiO₂ paste and dye-deposited blocking layer increases the conversion efficiency, charge collection efficiency, Hall carrier concentration, mobility, etc. Those enhancements were attributed to the increased carrier concentration, improved charge transfer, and diminished charge recombination due to the existence of graphene-functionalized interfaces. Perhaps the
The largest role of graphene materials as a DSSC sensitizer would be in a scenario which took advantage of the quantum effects and allowed for ultrafast injection in order to overcome intrinsic limitations to conventional devices.

3. Conclusion and Outlook

As a novel photovoltaic technology, dye-sensitized solar cells (DSSCs) have the potential to compete with traditional solar cells. Compared with silicon solar cells, they are insensitive to impurities in the fabrication process, which accelerates a transition from the research laboratory to the mass production line. Graphene, emerging as one the most unique materials, has been recently explored extensively in DSSCs to increase the efficiency and reduce the cost. Graphene has been widely explored in DSSCs as photoelectrodes. It shows great potential to change the solar industry in terms of financially and performance as well. Considering the photoelectrode, it is believed that a correct amount of graphene addition may really enhance the photocurrent in particular cases. Based on the reported results, it is very much evident that graphene materials will play a key role in enhancing the efficiency of the DSSCs. Nevertheless, as research progresses, we should always remember that graphene could have various properties according to their different manufacturing processes and each may be beneficial to different areas in a solar cell. A next stage of research, to bring graphene materials to a higher relevance in the DSSC community, would be to study whether improvements discussed within this review can be carried over to the current best-in-class devices.

Conflicts of Interest

The authors declare no competing financial interest.

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