Intrinsic ZnO/Al-doped ZnO Homojunction: Structural and Optical Properties

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ABSTRACT

Intrinsic zinc oxide (ZnO)/Al-doped ZnO (AZO) homojunction film was prepared using two-step immersion processes. The film was characterized using field emission scanning electron microscopy, X-ray diffraction (XRD), Raman spectroscopy, and ultraviolet–visible spectrophotometer to investigate their structural and optical properties. The surface morphology image displays that the ZnO deposited on the nanorod surfaces in layer form with average diameter of nanorods about 95 nm. The structural properties of XRD pattern demonstrate that the film possessed good crystallinity with the preferred orientation at (002) plane. The film also possessed excellent absorption in the ultraviolet (UV) region with optical band gap energy of 3.22 eV. These results indicate that the film has a good potential for optical-based device such a UV sensor.

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1. INTRODUCTION

Zinc oxide (ZnO) is an n-type which is used in various applications include sensors, solar cells, light emitting diodes, and transistors [1–4]. These wide applications are benefited from its properties such as wide bandgap energy, non-toxic, low cost, and also able to be produced into nanoscale structures. Previous studies reported that ZnO can be produced into various kinds of nanostructures such as nanowires, nanorods, nanospheres, nanoflowers, and nanoflakes. However, one-dimensional structures (nanorod and nanowires) are more promising in device fabrications due to ease of electron transfer and reducing grain boundary-related defect [5]. To date, there are lots of fabrication methods that are used to produce ZnO nanostructures such as metal organic chemical vapour deposition (MOCVD), sputtering, electrodeposition, thermal chemical vapour deposition, and hydrothermal [6]. Among the fabrication methods, water-based preparation is favourable due to simple process and low temperature use. Besides, this method also capable of producing high quality nanostructure films, comparable to the methods which use high temperature and high vacuum conditions.
Based on the previous studies, intrinsic ZnO coating, forming whether homojunction or heterojunction, manage to improve the properties of a film. According to Mamat et al., by coating ZnO nanorod arrays with intrinsic ZnO manage to improve the ultraviolet (UV) sensing performance due to the reduction of dark current [7]. In other study, Jiang et al. reported that the efficiency of CuIn\(_{1-x}\)Ga\(_x\)Se\(_2\) (CIGS) solar cells improved after coated with intrinsic ZnO [8]. In this study, we investigated the effect of ZnO/Al-doped ZnO (AZO) homojunction film in term of structural and optical properties. To the best of our knowledge, detail discussion on such homojunction film is rarely reported. The understanding on their properties is crucial for future implementation in devices.

2. RESEARCH METHOD

The preparation of ZnO/AZO film involved three steps. First is a seed layer preparation. The detail preparation of seed layer has been reported in previous study [9]. For the preparation of AZO, zinc nitrate hexahydrate (0.1 M), hexamethylenetetramine (0.1 M), and aluminum nitrate nonahydrate (0.001M) were used as the main materials. The reagents were combined and then going through sonication process for 30 min (50 °C). Then the solution was stirred for 3h, followed by immersion to grow the nanorod array structures. Finally, the film was annealed for 1 h (500 °C). Then, the second immersion for homojunction layer was conducted by mixing zinc nitrate hexahydrate (0.02 M) and hexamethylenetetramine (0.02M). Similar processes involving sonication, stirring, immersion, and annealing were used, where every parameters for each step were maintained. The illustration for the ZnO/AZO homojunction film preparation is depicted in Figure 1. For structural analyses, field emission scanning electron microscopy (FESEM, JEOL JSM-7600F), X-ray diffraction (PANalytical X’Pert PRO), and Raman spectroscopy (Horiba Jobin Yvon-79 DU420A-OE-325) measurements were conducted. For optical characterizations, ultraviolet–visible (UV–vis) spectrophotometer (Varian Cary 5000) was used.

![Diagram of ZnO/AZO homojunction film preparation](image)

Figure 1. The illustration for the ZnO/AZO homojunction film preparation

3. RESULTS AND ANALYSIS

The FESEM images in Figure 2 depict the surface morphology of the prepared samples. The images were taken at 100,000 x magnifications. Figure 2(a) shows the surface morphology of granular ZnO seed layer film. The average diameter of the grain is about 25 nm. Figure 2(b) demonstrates the AZO film grown on the ZnO seed layer film. The average diameter of the nanorods is about 70 nm. It is observed that the film consist of large pore areas in between the nanorods. Figure 2(c) depicts the ZnO film (0.02 M solution concentration) which was grown on the seed layer film. It is witnessed that only small part of the film is covered by the nanorods. Besides, the diameter of the nanorods is relatively small compared to AZO film, about 35 nm. Other than that, we also observed nanoflake structure appears on the film in addition to the nanorod structures. Figure 2(d) displays the ZnO/AZO homojunction film. Compared to the bare AZO film,
it is witnessed that the nanorod arrays of ZnO/AZO is covered by a slightly light color layer, which is intrinsic ZnO. The average diameter of the film is about 95 nm. Based on the image, it is indicates that the deposition of the intrinsic homojunction layer begin from the side surface, follows by the deposition on the top of the nanorods. This is based on the observation from the FESEM images that shows the less coverage of the intrinsic layer on the top surface of the nanorods. Interestingly, the growth process of the intrinsic ZnO does not develop into any form of nanostructures, but a layer. Such formation occurs may be due to the same crystal structure of ZnO and AZO, which enhances the homoepitaxy growth on the previous crystal (AZO).

Figure 2. Surface morphology of (a) seed layer, (b) AZO, (c) ZnO, and (d) ZnO/AZO films at 100,000 × magnifications

Figure 3 depicts the XRD pattern of the synthesized films. The diffraction angles measured are from 25° to 60°. The XRD pattern of each film is plotted individually and the intensity of the X-ray is shown in count-per-units (cps) to show the different in the intensity of each film. From the plots, there are several peaks occurred on the pattern, namely: (100), (002), (101), (102), and (110). For AZO film (Figure 3(a)), the dominant diffraction peak is at (002) plane orientation (along c-axis). For intrinsic ZnO film (Figure 3(b)), the dominant peak is at (100) and (101) plane orientations. Meanwhile, the growth of ZnO/AZO film also exhibits a preferential growth along c-axis (Figure 3(c)). In case of ZnO film, the preferential crystal growth may be dominated by the seed layer film and small number of nanoflake structures (from FESEM images), which preferentially grown along (100) and (101) plane orientations. Once the ZnO layer is grown on the AZO, we can observe that the peak intensity at (100) and (002) slightly increased. The increment of preferential a-plane growth may explain the formation of layer on the AZO surface which is expected to originate from the side surface of the nanorods of the AZO film as discussed earlier. Figure 3(d) depicts the magnified image of the XRD patterns at (002) plane orientations. The plot shows the significant increases of peak intensity for the film which was grown with AZO. Comparing both AZO and ZnO/AZO films, only a slight increment to the peak intensity can be detected. Interestingly, no peak shifting can be witnessed after the second growth process, indicating no residual stress forced on the ZnO/AZO film.

To investigate more on the crystal structure, we characterized the films using Raman spectroscopy as shown in Figure 4. The characterization was done under argon (Ar) laser operating at 514 nm as excitation source. Two inherent Raman peaks at 437 and 581 cm⁻¹ which correspond to E₂g(high) and E₁g(LO) mode, respectively [10]. Other insignificant peak can also be observed which situated at 332 cm⁻¹, which represents 2E₂g(M) mode [11]. E₂g(high) mode indicates the Raman vibration of hexagonal ZnO structure and the intensity of Raman vibration of E₂g(high) increased which shows that better crystalline structure of AZO and ZnO/AZO. This result is in agreement with the previously discussed XRD patterns. Further, the Raman vibration of E₁g(LO) mode shifted from 565 cm⁻¹ of seed layer and AZO to 581 cm⁻¹ of ZnO/AZO. E₁g(LO) mode can be related to the lattice defect (oxygen vacancy).
The optical properties of the films were measured using UV-vis spectrophotometer in the wavelength range of 350 – 800 nm. Figure 5 depicts the transmittance measurement of the films. All films exhibit good optical transparency in the visible region and steep edge around 380 nm, representing the band gap energy of the ZnO. The average transmittances of each film are 94.4%, 87.8%, and 66.5% for seed layer, AZO and ZnO/AZO homojunction films, respectively. The seed layer film possessed the lowest transmittance, which may be due to the lowest thickness compared to other films. Similar to other films, the variation of average transmittance is expected from the difference in film thicknesses, which is the cause for optical scattering effect [12].

Figure 3. XRD pattern of (a) AZO, (b) ZnO and (c) ZnO/AZO films

Figure 4. Raman spectra of (a) seed layer, (b) AZO and (c) ZnO/AZO films

Figure 5. Transmittance spectra of (a) seed layer, (b) AZO and (c) ZnO/AZO films
From the transmittance plot, the absorption coefficient of the films was plotted using the following Equation [13]:

\[ \alpha = \frac{1}{t} \ln \left( \frac{1}{T} \right) \] (1)

Here, \( \alpha \) is the absorption coefficient, \( t \) is the thickness, and \( T \) is the transmittance. The absorption coefficient is shown in Figure 6. The films show good absorption at UV region, with ZnO/AZO homojunction film possessed the highest UV absorption compared to other samples. The possible reason which leads to this result is high scattering effect at the homojunction interface which causes more photon trapping in the film.

![Figure 6. Absorption coefficient of (a) seed layer, (b) AZO and (c) ZnO/AZO films](image)

For direct band gap semiconductor, the optical band gap energy, \( E_g \) can be correlated through the given formula [14]:

\[ ahv = B(hv - E_g)^{\frac{1}{2}} \] (2)

Here, \( B \) is a constant and \( hv \) is the incident photon energy. Figure 7 demonstrates the \((ahv)^2\) versus photon energy for \( E_g \) estimation. The \( E_g \) of the seed layer, AZO and ZnO/AZO are 3.26, 3.27, and 3.22 eV, respectively. No significant changes to the \( E_g \) can be observed. However, one the interesting alteration to the \( E_g \) is when the intrinsic ZnO is used on the film. We can see that the \( E_g \) shifted to low value for the ZnO/AZO homojunction film. This alteration may be appeared due to the intrinsic nature of ZnO which has \( E_g \) value from 3.2 – 3.3 eV [15]. In addition to that, for ZnO/AZO film, the decrement of \( E_g \) may be caused by the scattering effect at the interface between the oxide homojunction [16].

![Figure 7. Tauc’s plot of (a) seed layer, (b) AZO and (c) ZnO/AZO films](image)
4. CONCLUSION

Intrinsic ZnO/Al-doped ZnO (AZO) homojunction film was prepared using two-step immersion processes. The FESEM images display that the intrinsic ZnO film depositions on the seed layer grew into ZnO nanostructures (nanorods and nanoflakes). However, when the intrinsic ZnO film is deposited on the nanorod surfaces, it just formed a layer which covered the nanorods of AZO film. The average diameter of the ZnO/AZO homojunction also increased up to 95 nm, higher than bare AZO. The XRD patterns demonstrate that the films possessed good crystallinity. AZO and ZnO/AZO homojunction films showed preferred orientation at (002) plane while ZnO film shows preferential growth along (100) and (101) plane orientations. Raman measurement confirms that the films consist of hexagonal ZnO structure with vibration peaks occurred at 2E(M), E_g(high), and E_g(LO). The ZnO/AZO homojunction film possessed excellent absorption at UV region with optical band gap energy of 3.22 eV. These results indicate that the ZnO/AZO homojunction film has a good potential for optical-based device such a UV sensor.

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REFERENCES

[1] Kokate SK, Supekar AT, Baviskar PK, Palve BM, Jadkar SR, Mohite KC, Pathan HM. CdS sensitized pristine and Cd doped ZnO solar cells: Effect of SILAR cycles on optical properties and efficiency. Materials Science in Semiconductor Processing. 2018; 80: 179-183.

[2] Voss T, Waldvogel SR. Hybrid LEDs based on ZnO nanowire structures. Materials Science in Semiconductor Processing. 2017; 69: 52-56.

[3] Ismail AS, Mamat MH, Yusoff MM, Malek MF, Zoolfakar AS, Rani RA, Suriani AB, Mohamed A, Ahmad MK, Rusop M. Enhanced humidity sensing performance using Sn-Doped ZnO nanorod Array/SnO2 nanowire heteronetwork fabricated via two-step solution immersion. Materials Letters. 2018; 210: 258-262.

[4] Jiang L, Huang K, Li J, Li S, Gao Y, Tang W, Guo X, Wang J, Mei T, Wang X. High carrier mobility low-voltage ZnO thin film transistors fabricated at a low temperature via solution processing. Ceramics International. 2018; 44(10): 11751-11756.

[5] Urgessa ZN, Botha JR, Tankio Djikop SR, Coleman C, Bhattacharyya S. Patterned growth of ZnO nanorods by chemical bath deposition. Physica B: Condensed Matter. 2018; 535: 79-83.

[6] Nava Núñez MY, Martínez-de la Cruz A. Nitric oxide removal by action of ZnO photocatalyst hydrothermally synthesized in presence of EDTA. Materials Science in Semiconductor Processing. 2018; 81: 94-101.

[7] Mamat MH, Malek MF, Hafizah NN, Md Sin ND, Saurdi I, Suriani A.B, Nafarizal N, Ahmad MK, Rouhi J, Rusop M. Fabrication of Intrinsic Zinc Oxide-Coated, Aluminium-Doped Zinc Oxide Nanorod Array-Based Ultraviolet Photoconductive Sensors. Applied Mechanics and Materials. 2015; 773-774: 696-700.

[8] Jiang M, Tang K, Yan X. Characterization of Intrinsic ZnO Thin Film Deposited by Sputtering and Its Effect on CuIn1-xGa_xSe2 Solar Cells. Journal of Photonics for Energy. 2012; 2(1): 28502.

[9] Mamat MH, Salhdan MZ, Khusaimi Z, Ahmed AZ, Abdullah S, Rusop M. Influence of doping concentrations on the aluminum doped zinc oxide thin films properties for ultraviolet photocoductive sensor applications. Optical Materials. 2010; 32(6): 699-699.

[10] Wang M, Ren F, Zhou J, Cai G, Cai L, Hu Y, Wang D, Liu Y, Guo L, Shen S. N Doping to ZnO Nanorods for Photoelectrochemical Water Splitting under Visible Light: Engineered Impurity Distribution and Terraced Band Structure. Sci. Rep. 2015; 5: 12925.

[11] Chanda A, Gupta S, Vasundhara M, Joshi SR, Mutta GR, Singh J. Study of structural, optical and magnetic properties of cobalt doped ZnO nanostructures. RSC Advances. 2017; 7(80): 50527-50536.

[12] Ismail AS, Mamat MH, Md. Sin ND, Malek MF, Zoolfakar AS, Suriani AB, Mohamed A, Ahmad MK, Rusop M. Fabrication of hierarchical Sn-doped ZnO nanorod arrays through sonicated sol-gel immersion for room temperature, resistive-type humidity sensor applications. Ceramics International. 2016; 42(8): 9785-9795.

[13] Lee S-M, Joo Y-H, Kim C-I. Influences of film thickness and annealing temperature on properties of sol-gel derived ZnO-SnO2 nanocomposite thin film. Applied Surface Science. 2014; 320: 494-501.

[14] Mohamed R, Mamat MH, Ismail AS, Malek MF, Zoolfakar AS, Khusaimi Z, Suriani AB, Mohamed A, Ahmad MK, Rusop M. Hierarchically assembled tin-doped zinc oxide nanorods using low-temperature immersion route for low temperature ethanol sensing. J Mater Sci: Mater Electron. 2017; 28(21): 16292-16305.

[15] Ahmed MAM, Mwankemwa BS, Carleschi E, Doyle BP, Meyer WE, Nel JM. Effect of Sm doping ZnO nanorods on structural optical and electrical properties of Schottky diodes prepared by chemical bath deposition. Materials Science in Semiconductor Processing. 2018; 79: 53-60.

[16] Yu S, Ding L, Zheng H, Xue C, Chen L, Zhang W. Electrical and photoelectric properties of transparent Li-doped ZnO/ZnO homojunctions by pulsed laser deposition. Thin Solid Films. 2013; 540: 146-149.