Polymorphism in magic-sized $\text{Au}_{144}(\text{SR})_{60}$ clusters

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Ultra-small, magic-sized metal nanoclusters represent an important new class of materials with properties between molecules and particles. However, their small size challenges the conventional methods for structure characterization. Here we present the structure of ultra-stable $\text{Au}_{144}(\text{SR})_{60}$ magic-sized nanoclusters obtained from atomic pair distribution function analysis of X-ray powder diffraction data. The study reveals structural polymorphism in these archetypal nanoclusters. In addition to confirming the theoretically predicted icosahedral-cored cluster, we also find samples with a truncated decahedral core structure, with some samples exhibiting a coexistence of both cluster structures. Although the clusters are monodisperse in size, structural diversity is apparent. The discovery of polymorphism may open up a new dimension in nanoscale engineering.

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The promise of nanotechnology, to engineer materials at the nanoscale with improved properties, is predicated on the idea that material structure and properties are fundamentally modified on this scale. Gold clusters are prototypical inorganic materials that exemplify this\(^1\)–\(^3\). In addition to being technologically important in their own right\(^4\), they are a model system for studying this paradigm, as they form ultra-stable 'magic number' molecule-like clusters of different sizes\(^10,11\). A major challenge, in the majority of cases where the clusters cannot be crystallized, is to determine their structure. We overcome this 'nanostructure problem'\(^12\) by using atomic pair distribution function (PDF) analysis of X-ray diffraction (XRD) data to study the structure of \(\text{Au}_{144}(\text{SR})_{60}\) (where R is the organic part of the thiol), one of the largest of the ultra-stable magic-sized clusters with known composition\(^13,14\). The PDF data successfully yield the core structure, with the surprising result that these clusters exhibit polymorphism. In very recent studies, single crystal structure determination illustrated that the much smaller \(\text{Au}_{14}(\text{SR})_{24}\) interconverts reversibly between two forms, depending on temperature\(^15\). Here we use PDF to show that polymorphism exists also in the large \(\text{Au}_{144}(\text{SR})_{60}\) cluster, representing the size regime in the transition between clusters forming non-bulk geometric structures and bulk face-centred cubic (fcc) nanoparticles\(^8\). The discovery of polymorphism brings an additional dimension to the phase space for nanoscale engineering.

The \(\text{Au}_{144}(\text{SR})_{60}\) structure has already been subject to many studies. Initially described as a ubiquitous 29-kDa core-mass compound\(^14,16,17\), more recently the composition was determined by mass spectrometry as \(\text{Au}_{144}(\text{SR})_{60}\) (refs \(^18,19\)). Lopez-Acevedo et al.\(^20\) developed a detailed structural model, tested by density functional theory (DFT), where the cluster consists of an icosahedral gold core surrounded by a gold/thiol surface layer. NMR (nuclear magnetic resonance) studies later suggested that all ligands are in symmetry equivalent positions\(^21\). Scanning transmission electron microscopy (STEM) studies by Bahena et al.\(^22\) were consistent with the icosahedral core and by introducing the NMR symmetry requirement in theoretical calculations they proposed a symmetrized structure model featuring an equivalent ligand arrangement\(^22\). This model consists of a gold core of 54 atoms arranged as two Mackay icosahedral shells (Fig. 1a), whereas a 60-atom layer covers the 55-site inner core in an ‘anti-Mackay’ manner (Fig. 1b). The surface of the cluster structures consist of -SR-Au-SR-type structures (Fig. 1d), referred to as ‘staples’\(^23\), and in combination this gives the full proposed structure as illustrated in Fig. 1c. The structure is closely related to that of \(\text{Pd}_{150}(\text{CO})(\text{PET})_{10}\) determined by single-crystal XRD\(^24\).

In this study, we apply atomic PDF analysis to \(\text{Au}_{144}(\text{SR})_{60}\). PDF analysis (described in Supplementary Note 1) has become widely used for nanostructure analysis\(^25–29\) and is a potential tool for nanocluster solution\(^30–32\). In recent times, PDF has also been applied to the fingerprinting of gold nanostructures\(^33,34\). PDF goes beyond conventional X-ray powder diffraction, which typically covers only a narrow range of reciprocal space\(^32\) and neglects diffuse scattering. The total scattering approach contains significantly more structural information, allowing a quantitative assessment of the structure that is impossible with conventional data from such small particles. We apply PDF nanostructure analysis to \(\text{Au}_{144}(\text{SR})_{60}\) clusters prepared with different ligands (Fig. 1e): phenylethane thiol (PET), para-mercaptobenzoic acid (p-MBA), butane thiol (SC4), hexanethiol (SC6) and dodecanethiol (SC12). Sample homogeneity is characterized by electrospray ionization–mass spectrometry (ESI–MS) and electrochemical methods. The approach results in full quantitative refinements of the structure of the gold core, with a semi-quantitative assessment of the surface structure. Surprisingly, we find two distinct structural forms for this cluster’s core, one based on icosahedra seen in smaller clusters, proposed earlier for this 144 gold atom cluster\(^20–22\), and one based on close packed decahedra that resemble larger gold clusters and bulk gold. The discovery of polymorphism in gold nanoclusters opens up a new dimension in nanoparticle engineering, presenting the

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**Figure 1 | Structure of the icosahedral \(\text{Au}_{144}(\text{SR})_{60}\) cluster.** (a) Fifty-four atom gold core consisting of two Mackay icosahedron shells. (b) The icosahedral gold core (pink) is covered by 60 gold atoms (yellow) making up the grand core. (c) Total structure, where the grand core is covered in ‘staples’—green atoms represent sulfur, whereas blue atoms represent gold in the staple structure. The organic carbon chains have been left out for clarity. (d) Illustration of staple structure on gold surface. (e) Thiolate ligands used in the study. From left: p-MBA, PET, SC4, SC6 and SC12.
Results
PDFs from Au_{144}(SR)_{60}. We first investigate the sample prepared with SC6 ligands. ESI–MS data (see Supplementary Fig. 1 and Supplementary Table 1) confirmed homogeneity of this sample, with at least 90% of the sample being Au_{144}(SC6)_{60} and a small byproduct (<10%) with ESI–MS peaks, which can be assigned to Au_{137}(SC6)_{56} (ref. 36). Previously in the literature, this impurity signal has been assigned to Au_{144}(SR)_{60} fragments37,38. No other cluster sizes, such as Au_{130}(SR)_{50} or Au_{133}(SR)_{52} were detected. The low Q scattering signal (where $Q = 4\pi \sin(\theta)/\lambda$ is the magnitude of the scattering vector), corresponding to conventional XRD data, the total scattering structure function $F(Q)$ and the PDF, $G(r)$, from this sample at 100 K are shown in Fig. 2. Owing to the small size of the gold clusters, only very broad scattering peaks are present in the low Q signal (Fig. 2a), resulting in too little information to attempt a total structure solution by crystallographic means. In the total scattering structure function $F(Q)$ (Fig. 2b) we see that the diffuse scattering extends over a wide range of reciprocal space containing scattering features with rich information that cannot be resolved when just the low Q conventional XRD data are used. The PDF, plotted in blue in Fig. 2c, is the Fourier transform of the data in Fig. 2b. This real-space function contains peaks at distances separating pairs of atoms in the structure. The observation of sharp peaks in real-space indicates that the gold clusters have a well-defined structure. The peaks in $G(r)$ disappear above 12.5 Å, which puts a lower bound on the diameter of the gold core of the clusters. The first large peak at ca. 2.9 Å is the nearest-neighbor gold–gold distance, $r_{NN}$, and is sharp. The strength and sharpness of the low $r$ peaks suggests a high multiplicity for these distances, indicating a rather well-packed structure. In Fig. 2c, we also show the experimental PDFs from the Au_{144}(PET)_{60} sample plotted in green. The similarity of the PDFs from Au_{144}(SC6)_{60} and Au_{144}(PET)_{60} indicates that these clusters have identical core structures and also establishes the reproducibility of the PDF measurements. The red line in Fig. 2c shows the PDF from Au_{144}(p-MBA)_{60}. There is a remarkable difference between this PDF and those of Au_{144}(SC6)_{60} and Au_{144}(PET). These clusters have the same size, as evident from the disappearance of sharp features in the PDF, by the characteristic, well-defined differential pulse voltammetry and from mobility in polycrylamide gel electrophoresis consistent with Au_{144}(SR)_{60}. Both the SC6, PET- and p-MBA-protected preparations, formed the poorly diffracting hexagonal plate crystals previously observed for these compounds39,40. The sharp PDF peaks indicate that the Au_{144}(p-MBA)_{60} clusters also have a well-defined ordered structure. However, their structure is remarkably different from that of the SC6- and PET-terminated clusters: the Au_{144}(SR)_{60} clusters are exhibiting polymorphism. We now explore quantitatively the two structural polymorphs, Form I and Form II, of these clusters.

Form I. We begin by calculating PDFs from candidate structures suggested in the literature to compare with the data. The relative atomic positions are highly constrained in the modelling (described in detail in Supplementary Note 2), with only five parameters allowed to vary: a scale factor accounting for the overall PDF intensity, a uniform cluster expansion factor that allows the cluster structure to contract or expand, two isotropic atomic displacement parameter applied separately to the core and surface atoms, as well as a parameter accounting for correlated atomic motion41. Therefore, good fits to the data are a strong indicator that the model has captured the correct geometry of the core. To simplify the models, only the Au and S atoms were included in the refinements, as the scattering signal from the organic ligands is negligible (see Supplementary Note 3). Figure 3a shows the calculated PDF from the model suggested by Bahena et al.42, fitted to the SC6 data20. The result of a refinement to the same data, but using the structure reported by Lopez-Acevedo et al.20 is shown in Supplementary Fig. 2. Both models describe the main features of the PDF of the Au_{144}(SC6)_{60} nanoclusters very well, with the Bahena structure giving a slightly better fit to the PDF with agreement factor $R_W = 16.3\%$. This confirms the proposed structures of previous theoretical and STEM studies41,42.
Similar fits to the Au$_{144}$(PET)$_{60}$ sample are given in Supplementary Fig. 3 and Supplementary Table 2, also showing good agreement with the icosahedral model ($R_W = 15.8\%$). Furthermore, a direct comparison of the experimental data from the PET and SC6 data show that the two samples give rise to practically identical PDFs as illustrated in Supplementary Fig. 4, where the difference curve between the two PDFs is essentially a flat line. Interestingly, the ESI–MS data indicated $B$ where the difference curve between the two PDFs is essentially the PET and SC6 data show that the two samples give rise to good agreement with the icosahedral model ($R_W = 7.8\%$, illustrating reproducibility of the synthesis and reliability of the measurements. We first attempt to use the Form I icosahedral structural model to establish whether this can be made to fit the different PDF by adjusting the refinement parameters. However, the model gives a very poor fit with a large difference between the calculated and measured PDF, and poor fit residue of $R_W = 36.0\%$ as shown in Fig. 3b. To further confirm that the sample does not simply contain stable clusters of a different size, for example, Au$_{144}$(SR)$_{54}$ (ref. 3), Au$_{130}$(SR)$_{50}$ (ref. 42) or Au$_{133}$(SR)$_{52}$ (ref. 43), we fitted known structural models for these clusters to the p-MBA data. In all cases, the models gave very poor agreements with the data (fits shown in Supplementary Figs 7–9), confirming that the samples are not made up of other stable cluster sizes.

Therefore, other models for the Au$_{144}$(SR)$_{60}$ gold cluster were explored. Initially, we considered only the positions of the 144 gold atoms and ignored the ligands in the model. We bascd this on the dominating scattering power of gold compared with the thiold, as discussed further in the Supplementary Note 3. First, a series of close-packed core models were constructed, closely related to bulk fcc gold. These included a 147-atom cuboctahedron, as well as clusters formed by cutting a sphere of ~144 atoms from fcc and hexagonal close-packed (hcp) lattices. The next attempted model was a two-phase fit of the PDFs from cutouts from fcc and hcp, which has been used as a proxy model in PDF modelling for close-packed structures that contain stacking faults$^{27}$. A summary of these simulations is given in Supplementary Table 3 and Supplementary Note 4. None of the fcc- or hcp-based clusters produced completely convincing fits to the observed PDF. However, the fits were significantly better than for the Bahena model, especially for the fcc/hcp model.
mixture as shown in Fig. 3c. The PDF agreement of the fcc/hcp model was remarkably improved after allowing for a separate expansion ratio for the atoms in the outermost shell, giving \( R_W = 16.3\% \), a step which was motivated by allowing for a possible surface relaxation. This improved the refinement by fitting the asymmetry in the first Au–Au peak. However, the results indicated that the bond lengths between the atoms in the surface were contracted compared with the bonds in the core and the atomic displacement parameters were excessively large over 0.03 Å² for the core atoms, suggesting the existence of some atomic relaxations that are not part of these simple models. Furthermore, this model contains 141 atoms in the fcc phase and 147 atoms in the hcp phase. We seek a model that can also explain the high stability of the Au core with 144 atoms, whereas spherical chunks of close-packed bulk material would not have special stability. Nonetheless, the fitting results establish that the structure of \( \text{Au}_{144}(\text{SR})_{60} \) is much closer to a three-dimensional close-packed structure than the icosahedral, DFT-derived models.

Our search for close-packed structures that have special atom counts led us to explore a series of Marks decahedral structures that are constructed by introducing twin boundaries along the (110) planes of the fcc lattice, as described in more detail in Supplementary Note 5. Closed shell, truncated decahedra can be constructed with a large range of discrete number of atoms, including 144, that is, the exact number of gold atoms in the cluster. This structure and the fit to the experimental PDF are shown in Supplementary Fig. 10, where excellent fits are seen. However, as described above, thiolate ligands are known to create –SR–Au–SR– or –SR–Au–SR–Au–SR– 'staples' on gold surfaces. The short staple, that is, –SR–Au–SR– is mainly seen on larger clusters, where the curvature is small, as would be the case in the \( \approx 2 \) nm \( \text{Au}_{144}(\text{SR})_{60} \) structure, and \( \text{Au}_{144}(\text{SR})_{60} \) may thus better be represented as \( \text{Au}_{114}(\text{SR})_{60} \). This pointed us towards a smaller decahedral structure as the core of the cluster, as illustrated in Fig. 4a, b. The cluster shown has exactly 114 gold atoms, leaving 30 gold atoms for the staples as required by the putative stoichiometry. The fit of this cluster to the data is shown in Fig. 3d and, as illustrated, the model very well describes the experimental PDF. All distinct sharp peaks up to 8 Å are reproduced and the fit remains very close even at higher \( r \)-values where the features are broader and less resolved. In studies of smaller clusters, it has been shown that although the core of the cluster is decahedral, a shell of gold atoms may be seen between the core structure and the staple layer. Therefore, we tried stripping down the decahedral structure to a yet smaller core and reattaching the atoms as 'caps' on the remaining structure. However, interestingly, any modification to the 114 atom decahedral core highly deteriorated the PDF fit, making lower symmetry structures unlikely. This makes us confident in a core structure based on the 114 atom decahedron, closely related to the ino decahedron described by Cleveland et al.

Various configurations of the staples on the 114-atom decahedral structure cluster were then considered, where one example is presented in Fig. 4c and other selected models are shown in Supplementary Fig. 11. The process of attaching the staples is described in detail in Supplementary Note 6. staples were placed on the (111) surfaces as previously seen; however, to accommodate all ligands to the structure in a physically sensible manner, staples were also attached to the (100) surfaces, although this motif has not yet been reported. Several different models were constructed, which all give comparably good fits to the data with \( R_W \) values of ca. 15–18%, with one example shown in Fig. 3e, where the presence of staples fit to the shoulder of the nearest neighbour Au–Au peak. The PDF refinements were somewhat sensitive to the staple attachment, as subtle differences between the features in the fitted PDF can be observed. However, based on PDF data alone we cannot determine the exact ligand arrangement and further studies combining total scattering with techniques sensitive to the ligand attachment are needed to determine the surface structure with full confidence.

Nevertheless, the PDF analysis clearly shows that the \( \text{Au}_{144}(\text{p-MBA})_{60} \) core takes a decahedral structure, unlike the \( \text{Au}_{144}(\text{SC6})_{60} \) and \( \text{Au}_{144}(\text{PET})_{60} \) samples described above. We call this second stable structure for \( \text{Au}_{144}(\text{SR})_{60} \) Form II. The decahedral structure fits well in the thiol stabilized gold cluster structure series. From single-crystal XRD of smaller clusters, a strong effect of ligand on internal structure and allowed nuclearity can be inferred, with close-packed and icosahedral structures both observed. For instance, \( \text{Au}_{36}(\text{PET})_{24} \) (ref. 47), \( \text{Au}_{36}(\text{PET})_{24} \) (ref. 48) and \( \text{Au}_{126}(\text{PET})_{24} \) (ref. 43) have been determined to have icosahedral cores. This is in contrast to \( \text{Au}_{18}(\text{SC6H11})_{24} \) (ref. 49), \( \text{Au}_{18}(\text{PET})_{24} \) (ref. 50) and \( \text{Au}_{18}(\text{p-MBA})_{14} \) (ref. 3), which have close-packed cores. A cuboctahedron-like structure (which also has closed-packed motifs) was seen for the \( \text{Au}_{40}(\text{p-MBA})_{12} \) cluster by advanced single-particle electron microscopy methods. It has furthermore been shown that substituting Au by Ag in ligand-free clusters containing ca. 312 metal atoms changes the structure from fcc to icosahedral.

### Form I and II coexistence

We next attempted to find a trend in ligand type for stabilizing the different structural forms and tested the effects of using linear thiol ligands of different length, namely SC4 and SC12, which we compare with the SC6 and PET samples. The PDF data for SC4, SC12 and PET are shown in Fig. 5a–c along with fits using the Form I model. The refined parameters are given in Supplementary Table 4, where the data from the hexane thiolated sample (SC6) and PET samples show good agreement with the icosahedral Form I model, and the SC4 and SC12 samples give much larger residuum values of 17.9% and 18.6%, respectively. Interestingly, the disagreement between data and model is particularly large around \( r \approx 5 \) Å, which is exactly
the position for one of the most dominating peaks in the decahedral PDF. ESI–MS data from both the SC4 and SC12 samples showed Au144(SR)60, as well as impurity peaks corresponding to Au137(SR)56 in quantities comparable to the PET-protected samples. No other clusters were seen. As the presence of Au137(SR)56 in the ESI–MS data did not affect the PDF fits to the PET-protected sample and as no other clusters are identified by ESI–MS, we can rule out that the disagreement is due to the presence of other cluster sizes.

In Fig. 5d, the experimentally derived PET PDF (in Form I) has been subtracted from the SC12 calculated PDF and the difference curve is plotted below. Close inspection indicates that it strongly resembles the PDF of the Form II decahedral core structure, as seen when comparing with the calculated PDF from the 114-atom decahedron plotted along with the data. The difference curve has exactly the same features as seen from the decahedron phase, showing that the sample contains clusters of two distinct structures: Form I and Form II. Similar results are seen for the SC4 sample as illustrated in Supplementary Fig. 12. Two-phase fits showed that the SC4 sample contains 12% decahedral clusters (88% icosahedral clusters), whereas the SC12 sample has 14% decahedral clusters, as listed in Supplementary Table 5 and illustrated in Supplementary Fig. 13. When including the decahedral phase in the fit, the resulting R-values are reduced to ca. 16%. The results unambiguously show that two polymorphs of the Au144(SR)60 cluster are present.

Discussion
The question remains which factors affect the polymorph. Previous studies of gold clusters have indicated that ligand length may influence the structure of the gold core. Our total scattering data cannot confirm this trend, as both the longest (SC12) and shortest (SC4) linear ligand give mainly icosahedral clusters, with a smaller fraction of decahedral clusters present in each sample. The fact that we see both the icosahedral and decahedral clusters in samples made with the same ligands illustrate that the structural diversity is not a simple effect of the ligand chain length, bulkiness or bonding strength. It is a clear indication that the two structures are very close in energy. As discussed above, Wong et al. reported 1H-NMR studies of Au144(p-MBA)60 clusters, which showed only one doublet in the aromatic region of the spectrum, suggesting that all ligands are in symmetry equivalent positions. Interestingly, 13C-NMR studies on 29 kDa gold clusters have shown that the NMR signal is highly dependent on the charge state of the nanocluster. The simple NMR signal indicating symmetry equivalent ligands was seen only when the clusters were in charge state +3, whereas other signals were seen at lower charge states. In recent work, Tlahuice-Floret et al. used DFT to study the effect of charge state on structure of the gold subhalide Au144Cl60, which is isoelectronic with Au144(SR)60. It was shown that a fully symmetric icosahedral structure is stable at charge states +2 and +4, but not at neutrality. Furthermore, other studies have experimentally illustrated charge-dependent thermal stability of Au144(SC6H13)55. All our data have been measured in the uncharged state and, therefore, we cannot comment on a charge-dependent structure in Au144(p-MBA). However, when considering our new PDF data, applicable for detailed nanostructure analysis, along with the previously published powder x-Ray diffraction (PXRD), STEM and NMR data, this again points to a scenario where the icosahedral core structure and decahedral structure both exist with very similar energies. Small differences in the electronic state of the cluster from, for example, charge or ligand binding, could lead to different structures and, possibly, even switching between the different structural forms.

As noted above, the full staple arrangement on the decahedral clusters cannot be deduced from the PDFs and X-ray scattering data must be combined with techniques sensitive to the organic ligands to establish the total structure. If considering also the experimental PDFs from the icosahedral structures (that is, with SC6 and PET ligands), we note that neither the Bahena et al. nor Lopez-Acevedo et al. models fully capture the details in the PDFs, for example, in the peaks between 4 and 5 Å in Fig. 3a. Some structural details exist, which are not present in the established models. Therefore, further studies of the structure are needed, where scattering is combined with theory and spectroscopy, to establish the total structure.
The polymorphism seen in our data suggests many new studies of gold nanoclusters. The close energies between different cluster structures may not be limited to the Au$_{144}$(SR)$_{60}$ cluster family, but exist in a larger size range and in different materials systems. The presence of polymorphism challenges some of the characterization methods that are used for structure solution. For example, when applying single-crystal XRD, the crystallization process works as a structural sieve that will favour only one cluster polymorph, indicating that the energy of the two structures are very close to each other. In recent times, several new metal clusters coexist, indicating that the energy of the two structures are very close to each other. In recent times, several new metal clusters have been isolated in the size range from 50 to 300 atoms$^{17,18}$. The structures of many of these clusters remain undetermined, owing to difficulties in crystallizing the clusters into a large, single crystal suitable for structure determination. We believe that PDF will be an excellent tool for these studies and, when combined with spectroscopic methods, will be able to provide full structure solutions to many new nanomaterials.

**Methods**

**Synthesis and purification of Au$_{144}$(SR)$_{60}$**. Au$_{144}$(p-MBA)$_{60}$ was synthesized as in Ackerson et al.$^{39}$ and described in Supplementary Note 7, along with the purification and characterization. Au$_{144}$(SR)$_{60}$, where SR = PET, S4, S6 and S12, synthesis followed Qian et al.$^{39}$, also discussed further in Supplementary Note 7. The samples were characterized by ESI–MS (Supplementary Fig. 1 and Supplementary Table 1) and electrochemical methods (Supplementary Fig. 14–16).

**X-ray total scattering experiments**. Total scattering data were acquired during three different beamtimes at three different facilities. For all samples, the cluster powders were loaded in Kapton tubes with inner diameter of 1 mm. Data for both samples of the Au$_{144}$(p-MBA)$_{60}$ cluster was obtained at ID11 at the European Synchrotron Radiation Facility with an X-ray wavelength of 0.1774 Å at 100 K. For the Au$_{144}$(PET)$_{60}$, Au$_{144}$(6C6h$_{60}$), Au$_{144}$(SC4)$_{60}$ and Au$_{144}$(SC12)$_{60}$ clusters, data were measured at beamline 11-ID-B at the Advanced Photon Source, at Argonne National Laboratory. Here, data were measured at 100 and 300 K with X-ray wavelength 0.143 Å. Additional data sets for the Au$_{144}$(PET)$_{60}$ were furthermore measured at the X7B beamline at room temperature with X-ray wavelength of 0.319 Å, as well as at the X17A beamline, X-ray wavelength 0.186 Å at 100 and 300 K, both at the National synchrotron light source facility at Brookhaven National Laboratory. The experimental powder diffraction patterns were integrated using the software Fit2D$^{29}$ and Fourier transformed to obtain the PDF using the program PDFgetX$^{28}$. Modelling was done using DiffrX-MIL. Details on the PDF analysis can be found in Supplementary Note 1.

**Data availability**. The X-ray total scattering data that support the findings of this study are available at https://github.com/sbillinge/ncomms-goldnp-2016 with the 10.5281/zenodo.51551 (ref. 62).

**References**

1. Brust, M., Fink, J., Bethell, D., Schiffrin, D. J. & Kiely, C. Synthesis and reactions of functionalized gold nanoparticles. *J. Chem. Soc. Chem. Commun.* 1655–1656 (1994).
2. Brust, M., Walker, M., Bethell, D., Schiffrin, D. J. & Whyman, R. Synthesis of thiol-derivatized gold nanoparticles in a 2-phase liquid-liquid system. *J. Chem. Soc. Chem. Commun.* 801–802 (1994).
3. Jadzinsky, P. D., Calero, G., Ackerson, C. J., Bushnell, D. A. & Kornberg, R. D. Structure of a thiol monolayer-protected gold nanoparticle at 1.1 Ångstrom resolution. *Science* **318**, 430–433 (2007).
4. Walter, M. et al. A unified view of ligand-protected gold clusters as superatom complexes. *Proc. Natl Acad. Sci. USA* **105**, 9157–9162 (2008).
5. Zhu, M., Aiken, C. M., Hollander, F. J., Schatz, G. C. & Jin, R. Correlating the crystal structure of A thiol-protected Au-25 cluster and optical properties. *J. Am. Chem. Soc.* **130**, 5883–5889 (2008).
6. Weisser, H. C. et al. Information on quantum states pervades the visible spectrum of the ubiquitous Au-144(SR)(60) gold nanocluster. *Nat. Commun.* **5**, 3785 (2014).
7. Philip, R., Chantarassupawong, P., Qian, H. F., Jin, R. C. & Thomas, J. Evolution of nonlinear optical properties: from gold atomic clusters to plasmonic nanocrystals. *Nanos Lett.* **12**, 4661–4667 (2012).
8. Negishi, Y. et al. A critical size for emergence of nonbulk electronic and geometric structures in dodecanethiolate-protected Au clusters. *J. Am. Chem. Soc.* **136**, 1206–1212 (2014).
9. Daniel, M. C. & Asatru, D. Gold nanoparticles: assembly, supramolecular chemistry, quantum-size-related properties, and applications toward biology, catalysis, and nanotechnology. *Chem. Rev.* **104**, 293–346 (2004).
10. Jin, R. C. Quantum sized, thiolate-protected gold nanoclusters. *Nanoscale* **2**, 343–362 (2010).
11. Qian, H. F., Zhu, M. Z., Wu, Z. K. & Jin, R. C. Quantum sized gold nanoclusters with atomic precision. *Acc. Chem. Res.* **45**, 1470–1479 (2012).
12. Billinge, S. J. L. & Levin, I. The problem with determining atomic structure at the nanoscale. *Science* **316**, 1698–1697 (2012).
13. Whetten, R. L. et al. Nanocrystal gold molecules. *Adv. Mater.* **8**, 428–433 (1996).
14. Schaaff, T. G., Shafigullin, M. N., Khoury, J. T., Vezmar, I. & Whetten, R. L. Properties of a ubiquitous 29 kDa Au: SR cluster compound. *J. Phys. Chem. B* **105**, 8785–8796 (2001).
15. Tian, S. et al. Structural isomerism in gold nanoparticles revealed by X-ray crystallography. *Nat. Commun.* **6**, 8667 (2015).
16. Alvarez, M. M. et al. Critical sizes in the growth of Au clusters. *Chem. Phys. Lett.* **266**, 91–98 (1997).
17. Cleveland, C. L. et al. Structural evolution of smaller gold nanocrystals: the truncated decahedral motif. *Phys. Rev. Lett.* **79**, 1873–1876 (1997).
18. Chaki, N. K., Negishi, Y., Tsunoyama, H., Shichibu, Y. & Tsukuda, T. Ubiquitous 8 and 29 kDa gold: alkanethiolate cluster compounds: mass-spectrometric determination of molecular formulas and structural implications. *J. Am. Chem. Soc.* **130**, 8608–8612 (2008).
19. Qian, H. F. & Jin, R. C. Controlling nanoparticles with atomic precision: the case of Au-144(SCH2CH2Ph)(60). *Nanos Lett.* **9**, 4083–4087 (2009).
20. Lopez-Acevedo, O., Akola, J., Whetten, R. L., Gronbeck, H. & Hakkinen, H. Structure and bonding in the ubiquitous icosahedral metallic gold cluster Au-144(SR)(60). *J. Phys. Chem. C* **113**, 5035–5038 (2009).
21. Wong, O. A., Heinecke, C. L., Simone, A. R., Whetten, R. L. & Ackerson, C. J. Ligand symmetry-equivalence on thiolate protected gold nanoclusters determined by NMR spectroscopy. *Nanoscale* **4**, 4099–4102 (2012).
22. Rahena, D. et al. STEM electron diffraction and high-resolution images used in the determination of the crystal structure of the Au-144(SR)(60) cluster. *J. Phys. Chem. Lett.* **4**, 975–981 (2013).
23. Hakkinen, H. The gold-sulfur interface at the nanoscale. *Nat. Chem.* **4**, 443–455 (2012).
24. Tran, N. T., Powell, D. R. & Dahl, L. F. Nanosized Pd-145(CO)(x) (Pd3) containing a capped three-shell 145-atom metal-core geometry of pseudo icosahedral symmetry. *Angew. Chem. Int. Ed.* **39**, 4121–4125 (2000).
25. Billyng, S. J. L. & Kanatzidis, M. G. Beyond crystallography: the study of disorder, nanocrystallinity and crystallographically challenged materials with pair distribution functions. *Chem. Commun.* **7**, 749–760 (2004).
26. Young, C. A. & Goodwin, A. L. Applications of pair distribution function methods to contemporary problems in materials chemistry. *J. Mater. Chem.* **21**, 4646–4676 (2011).
27. Yang, X. H. et al. Confirmation of disordered structure of ultrasmall CdSe nanoparticles from X-ray atomic pair distribution function analysis. *Phys. Chem. Chem. Phys.* **15**, 8480–8486 (2013).
28. Neder, R. B. & Korsukov, V. I. Structure of nanoparticles from powder diffraction data using the pair distribution function. *J. Phys. Condens. Matter* **17**, S125–S134 (2005).
29. Page, K., Hood, T. C., Proffen, T. & Neder, R. B. Building and refining complete nanoparticle structures with total scattering data. *J. Appl. Crystallogr.* **44**, 327–336 (2011).
30. Juhas, P., Cherba, D. M., Duxbury, P. M., Punch, W. F. & Billinge, S. J. L. Ab initio determination of solid-state nanostucture. *Nature* **440**, 655–658 (2006).
31. Cliffe, M. J., Dove, M. T., Drabold, D. A. & Goodwin, A. L. Structure determination of disordered materials from diffraction data. Phys. Rev. Lett. 104, 125501 (2010).
32. Kumara, C., Zuo, X. B., Ilavsky, J., Cullen, D. A. & Dass, A. Atomic structure of Au-329(SR)(84) faradare plasma mononanocules. J. Phys. Chem. C 119, 11260–11266 (2015).
33. Kumara, C., Zuo, X. B., Dass, A. Faradare-940: synthesis, mass spectrometry, electron microscopy, high-energy X-ray diffraction, and X-ray scattering study of Au-940 + 20(SR) ~ 160 + 4 nanocrystals. ACS Nano 8, 6431–6439 (2014).
34. Kumara, C. et al. Super-stable, highly monodisperse plasma Faradare-500 nanocrystals with 500 gold atoms: Au-500(SR)(~120). J. Am. Chem. Soc. 136, 7410–7417 (2014).
35. Billinge, S. J. L. et al. Characterisation of amorphous and nanocrystalline molecular materials by total scattering. CrystEngComm 12, 1366–1368 (2010).
36. Jupally, V. R. et al. Au-137(SR)(56) nanomolecules: composition, optical spectroscopy, electrochemistry and electrocatalytic reduction of CO2. Chem. Commun. 50, 9895–9898 (2014).
37. Dharmaratne, A. C. & Dass, A. Au144-xCu(x)SC6H13(60) nanomolecules: efficacy of Cu coordination on composition and plasmon-like peak emergence in optical spectra. Chem. Commun. 50, 1722–1724 (2014).
38. Salorinne, K. et al. Nondestructive size determination of thiol-stabilized gold nanoclusters in solution by diffusion limited NMR spectroscopy. Anal. Chem. 85, 3489–3492 (2013).
39. Ackerson, C. J., Jadzinsky, P. D., Sexton, J. Z., Bushnell, D. A. & Kornberg, R. D. Sequence and biochemical properties of 2 and 3 nanometer gold nanoparticles. Bioconjug. Chem. 21, 214–218 (2010).
40. Koivisto, J. et al. Vibrational perturbations and ligand coupling in a single crystal of Au-144(SCH2Ph)(60) nanocluster. J. Phys. Chem. Lett. 5, 387–392 (2014).
41. Jeong, I. K., Proffen, T., Mohiuddin-Jacobs, F. & Billinge, S. J. L. Measuring correlated spin motion using X-ray diffraction. J. Phys. Chem. A 103, 921–924 (1999).
42. Chen, Y. X. et al. Crystal structure of barrel-shaped chiralAu-130(p-MBT)(50) nanocluster. J. Am. Chem. Soc. 137, 10076–10079 (2015).
43. Dass, A. et al. Au-133(SPh-Bu)(52) nanomolecules: X-ray crystallography, optical, electrochemical, and theoretical analysis. J. Am. Chem. Soc. 137, 4610–4613 (2015).
44. Marks, L. D. Surface structure and energetics of multiply twinned particles. Philos. Mag. 49, 81–93 (1984).
45. Marks, L. D. Experimental studies of small-particle structures. Rep. Prog. Phys. 57, 603–649 (1994).
46. Jian, D. E., Tiago, M. L., Luo, W. D. & Dai, S. The ‘Stapel’ motif: a key to stability of thiolate-protected gold nanoclusters. J. Am. Chem. Soc. 130, 2777–2784 (2008).
47. Heaven, M. W., Dass, A., White, P. S., Holt, K. M. & Murray, R. W. Crystal structure of the gold nanoparticle [N(CH2Ph)(7)](4)[Au-25(SCH2CH2Ph)(18)](18). J. Am. Chem. Soc. 130, 3754–3762 (2008).
48. Qian, H. F., Eckenhoff, W. T., Zhu, Y., Pintauer, T. & Jin, R. C. Total structure determination of thiolate-protected Au-38 nanoclusters. J. Am. Chem. Soc. 132, 8280–8284 (2010).
49. Das, A. et al. Structure determination of [Au-18(SR)(14)]. Angew. Chem. Int. Ed. 54, 3140–3144 (2015).
50. Das, A. et al. Cyclopentanehexadithio-protected Au-36(SC5H9)(24) nanocluster: crystal structure and implications for the steric and electronic effects of ligand. J. Phys. Chem. A 118, 8264–8269 (2014).
51. Azbel, M. et al. Electron microscopy of gold nanoparticles at atomic resolution. Science 345, 909–912 (2014).
52. Tian, N. & Palmer, R. E. Variation of the core atomic structure of thiolated (Ag, Au, Cu, Zn) nanoclusters with composition from aberration-corrected HAADF STEM. J. Phys. Chem. C 119, 11114–11119 (2015).
53. Song, Y., Harper, A. S. & Murray, R. W. Ligand heterogeneity on monolayer-protected gold clusters. Langmuir 21, 5492–5500 (2005).
54. Thaluque-Flores, A., Black, D. M., Bach, S. B. H., Jose-Yacamán, M. & Whetten, R. L. Structure & bonding of the gold-sulphide cluster {[Au144(36)]}. Phys. Chem. Chem. Phys. 15, 19191–19195 (2013).
55. Tofanelli, M. A. & Ackerson, C. J. Superatom electron configuration predicts thermal stability of Au-25(SR)(18) nanoclusters. J. Am. Chem. Soc. 134, 16937–16940 (2012).
56. Yamazoe, S. et al. Hierarchy of bond stiffnesses within icosahedral-based gold clusters protected by thiokolates. Nat. Commun. 7, 10414 (2016).
57. Li, G., Zeng, C. J. & Jin, R. C. Thermally robust Au-99(SPh)(42) nanoclusters for chemoselective hydrogenation of nitrobenzaldehyde derivatives in water. J. Am. Chem. Soc. 136, 3673–3679 (2014).
58. Nimimala, P. R., Yoon, B., Whetten, R. L., Landman, U. & Dass, A. Au-67(SR)(35) nanomolecules: characteristic size-specific optical, electrochemical, structural properties and first-principles theoretical analysis. J. Phys. Chem. A 117, 504–517 (2013).
59. Qian, H. F. & Jin, R. C. Ambient synthesis of Au-144(SR)(60) nanoclusters in methanol. Chem. Mater. 23, 2209–2217 (2011).
60. Hammersley, A. P., Svensson, S. O., Hartland, M., Fitch, A. N. & Hausermann, D. Two-dimensional detector software: from real detector to idealised image or two-theta scan. High Pressure Res. 14, 235–248 (1996).
61. Juhás, P., Davis, T., Farrow, C. L. & Billinge, S. J. L. PDFgetX3: a rapid and highly automatable program for processing powder diffraction data into total scattering pair distribution functions. J. Appl. Crystallogr. 46, 560–566 (2013).
62. Jensen, K. M. Ø. et al. Data for polymorphism in magic sized Au144(SR)(60) clusters. https://doi.org/10.5281/zenodo.51551 (2016).

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Author contributions
M.A.T., C.L.H. and C.H.A. synthesized and characterized the samples. K.M.Ø.J., S.J.L.B and G.V. collected the total scattering data. K.M.Ø.J. and P.J. analysed and modelled the data. K.M.Ø.J., P.J. and S.J.L.B. wrote the manuscript.

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