Impact of Substrates on the Electronic and Mechanical Properties of Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} Alloys

O. A. Alfrnwani\textsuperscript{1} · A. R. Degheidy\textsuperscript{1} · Elkenany B. Elkenany\textsuperscript{1}

Received: 6 December 2021 / Accepted: 23 February 2022 / Published online: 21 March 2022

Abstract
The compositional dependence of electronic and mechanical properties of a Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} quaternary alloy in zinc-blende structure lattice-matched to GaSb, InAs, and InP substrates is studied. The calculations are done based on the empirical pseudopotential method modified with virtual crystal approximation. Our calculations are obtained for the energy band gaps, elastic constants, elastic moduli, bond stretching, bond bending forces, and internal strain parameter. The material system of interest is found to be a direct semiconductor within a small range of Al concentrations about 0–0.07 and an indirect one outside this region. The Debye temperature and the Fröhlich coupling constant have been determined at different values of composition lattice-matched to different substrates. The phonon frequencies and the sound velocity for different substrates at various compositions have been studied. There is a good agreement between our results and the experimental data for its constituent binary compounds, AlP, AlSb, InP, InSb, and ternary alloys, InPSb, AlPSb, which supports the calculated results of the studied Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} quaternary alloy. The studied properties for the considered alloy may be helpful for the fabrication of optoelectronic devices.

Keywords Electronic properties · mechanical properties · Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} · different substrate · alloys

Introduction
Semiconductors consisting of the third and fifth group elements of the periodic table are the key materials for optoelectronic devices. Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} quaternary alloys are important materials and have important technological device applications.\textsuperscript{1} x and y are the concentrations of Al and P in the considered alloy, respectively. One advantage of the Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} alloy is that it can be grown lattice-matched to binary substrates, GaSb, InAs, and InP. Special attention has been given to the mechanical properties of semiconductors because they are one of the best tools for guiding the successful design and fabrication of optoelectronic devices.\textsuperscript{1–6} This work aims to focus on the effect of the used substrates (GaSb, InAs, and InP) on the electronic and mechanical properties in Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} under the effect of alloy composition. The studied quaternary alloy is bordered by the ternary alloys, Al\textsubscript{x}In\textsubscript{1−x}P, Al\textsubscript{x}In\textsubscript{1−x}Sb, AlP\textsubscript{y}Sb\textsubscript{1−x}, and InP\textsubscript{y}Sb\textsubscript{1−x}, which are bounded also by four binary compounds, AlP, AlSb, InP, and InSb. These binary compounds and ternary alloys were studied by Degheidy et al. and have been published in Refs. 7–10 The structural, electronic, mechanical, thermal, and optical properties of ternary and quaternary semiconductor alloys were investigated using different methods such as first-principles plane-wave method within the LDA approximations, EPM, and density functional theory (DFT).\textsuperscript{9–21}

In the present work, the direct and indirect energy band gaps, the elastic constants (C\textsubscript{11}, C\textsubscript{12}, C\textsubscript{44}), elastic moduli (B\textsubscript{0}, C\textsubscript{s}, Y\textsubscript{0}), and some related properties such as bond stretching (α), bond-bending (β) force constants and internal strain parameter (ξ) of Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} lattice-matched to different substrates have been studied. Our calculations are based on the empirical pseudopotential method (EPM) modified with virtual crystal approximation (VCA).\textsuperscript{22,23} Our results over the compositional range of Al concentration (x=0–1) in the Al\textsubscript{x}In\textsubscript{1−x}P\textsubscript{y}Sb\textsubscript{1−y} system are performed and compared with the published data in the literature and showed good agreement especially for binary compounds and ternary alloys which supports the results of the quaternary alloy of interest.

\textsuperscript{1} Department of Physics, Faculty of Science, Mansoura University, P. O. Box 35516, Mansoura, Egypt

\textsuperscript{2} Elkenany B. Elkenany
kena@mans.edu.eg
Computational Method

The present calculations were performed by using the empirical pseudopotential method (EPM) combined with the virtual crystal approximation (VCA). The fundamental benefit of employing pseudopotentials is that only valence electrons must be taken into account. The electrons in the core are handled as though they were frozen in an atomic-like state. The valence electrons are assumed to move in a weak one-electron potential as a result of this. Model pseudopotentials are substituted for actual pseudopotentials to further simplify the problem. Even though there are alternative methods for estimating electronic band structures, the pseudopotential method produces surprisingly accurate results for the amount of computer time and effort required. The EPM involves empirical parameters, namely the symmetric and anti-symmetric form factors which are modified to fit the experimental energy band gaps of the parent binary compounds of the studied alloy, namely AlP, AlSb, InP, and InSb. The obtained accurate form factors allow us to calculate the form factors of the quaternary Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ alloy of interest, where

$$ W_{\text{alloy}}(x) = xy W_{\text{AlP}} + y(1-x) W_{\text{InP}} + x(1-y) W_{\text{AlSb}} + (1-x)(1-y) W_{\text{InSb}} $$ (1)

The lattice constant of the Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ alloy can be also determined in terms of those of lattice constants of the constituent binary compounds $a_{\text{AlP}}$, $a_{\text{InP}}$, $a_{\text{AlSb}}$ and $a_{\text{InSb}}$ using Vegard's law as

$$ a_{\text{alloy}} = xy a_{\text{AlP}} + y(1-x) a_{\text{InP}} + x(1-y) a_{\text{AlSb}} + (1-x)(1-y) a_{\text{InSb}} $$ (2)

The Lyddane–Sachs–Teller relation were used to determine the longitudinal and transverse phonon frequencies (LO and TO). Knowing the form factors and the lattice constants of the binary compounds, where

$$ \frac{\omega_{\text{TO}}^2}{\omega_{\text{LO}}^2} = \frac{\varepsilon_{\infty}}{\varepsilon_0} $$ (3)

$$ \omega_{\text{LO}}^2 - \omega_{\text{TO}}^2 = \frac{4\pi e^2}{M \varepsilon_0} $$ (4)

where $\varepsilon_0$, $\varepsilon_{\infty}$, $\varepsilon_k$, $e$, $M$, $\Omega$ are the high-frequency dielectric constant, static dielectric constant, transverse effective charge, electron charge, twice the reduced mass, and volume occupied one atom, respectively.

The sound velocity was calculated using the crystal density ($g$) and stiffness constants ($c_{ij}$). If the quaternary alloy of interest is lattice-matched to a substrate, namely GaSb, InAs, InP, the lattice matching condition is obtained by using the lattice constant of a substrate instead of the lattice constant of the studied alloy. The substrate lattice matching relations are given by

For GaSb substrate

$$ y = 0.3835 - 0.3439 x/0.6104 + 0.0616 x^2, 0 \leq x \leq 1 $$ (6)

For InP substrate

$$ y = 0.6104 - 0.3439 x/0.6104 + 0.0616 x^2, 0 \leq x \leq 1 $$ (7)

For InAs substrate

$$ y = 0.4211 - 0.3439 x/0.6104 + 0.0616 x^2, 0 \leq x \leq 1 $$ (8)

Or in general,

$$ x = \frac{d_1 + d_3 y}{d_3 + d_4 y}, 0 \leq x \leq 1 $$ (9)

where $d_i$ are expressed in terms of the corresponding lattice constants of the binary compounds, where

$$ d_1 = a_{\text{sub}} - a_{\text{InSb}} $$ (10)

$$ d_2 = a_{\text{InSb}} - a_{\text{InP}} $$ (11)

$$ d_3 = a_{\text{AlSb}} - a_{\text{InSb}} $$ (12)

$$ d_4 = a_{\text{AlP}} + a_{\text{InSb}} - (a_{\text{InP}} + a_{\text{AlSb}}) $$ (13)

Knowing the form factors and the lattice constants of the Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ system for a certain composition parameter $x$, the energy eigenvalues are calculated by solving numerically the secular determinant, more details are found in

$$ \frac{1}{2} \left| k + \bar{G} \right|^2 - E(x) \left| k - \bar{G} \right|^2 + \sum_{\bar{G} \neq \bar{G}} V_{\bar{G}} \left( \Delta \bar{G} \right) = 0 $$ (14)

The band structure calculation in this method is designed to solve the eigenvalue problem for the energy $E$, as indicated by Eq. 14. $V(G')$ is the Fourier component of the periodic part of the Bloch function, as seen in Eq. 14. The size of the matrix and the precision of the calculation are both determined by the number of employed reciprocal lattice vectors. A specimen of length 50a is used to select $k(k_1, k_2, k_3)$ values, where the number of sample points is

---

Impact of Substrates on the Electronic and Mechanical Properties of Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$… (4151)

© Springer
twice the value of 50, i.e., 100, by taking into account the entire Brillouin zone (BZ) interval \([-1, 1]\). \(x\) is calculated for the energy band gaps for the binary compounds, AlP, AlSb, InP, and InSb, and some of their related alloys lattice-matched to GaSb, InAs, and InP for some Al concentrations and compared with the available published data.

## Results and Discussion

In Table I, we give our results for the calculated direct \(E^G_\Gamma(x)\), and indirect \(E^I_\Gamma(x), E^X_\Gamma(x)\) energy band gaps of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) lattice-matched to GaSb, InAs, and InP substrates for the compositional parameter, \(x\). Some results are compared with the available theoretical or experimental data from the literature\(^{1,8,9,36,39-41}\) and showed good agreement. It is known that the lattice constant of InP is less than that of InAs and GaSb. Because the energy band gaps and the lattice constant have an inverse relationship in this scenario, the energy band gaps of the quaternary alloy lattice-matched to InP are larger than those lattice-matched to GaSb and InAs substrates over the entire composition region.

The variation of the direct energy band gaps \(E^G_\Gamma\) of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) alloy versus its Al content for the three different substrates is displayed in Fig. 1. In this figure, we observe that \(E^G_\Gamma\) are increased with increasing the composition parameter \(x\) in the region 0–0.3 and then slightly decreased whatever substrate was used. The energy band gaps of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) lattice-matched to InP have larger values than those lattice-matched to GaSb and InAs substrates. However, the variation of \(E^G_\Gamma\) band gap of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) lattice-matched to GaSb and InAs substrates is very close.

Table II shows our results for the polarity and the transverse effective charge of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) lattice-matched to GaSb, InAs, and InP substrates. For comparison, available published data are presented and a good agreement is reported. With increasing Al concentration, \(x\), the polarity of \(Al_{x}In_{1-x}P_{y}Sb_{1-y}\) is slightly decreased for all used substrates.

### Table I: Band-gap energies of AlP, AlSb, InP, and their alloys lattice-matched to different substrates for some compositions (x) at room temperature and normal pressure

| Compound                             | Band gap energy (eV) |
|--------------------------------------|----------------------|
|                                      | \(E^G_\Gamma\) | \(E^L_\Gamma\) | \(E^X_\Gamma\) |
| InP, InSb, GaSb, and InAs            | Error %    | Error %    | Error %    |
| AlP                                  | 3.54       | 3.53       | 2.48       |
| AlSb                                 | 3.55\(^c\) | 3.53\(^c\) | 2.48\(^c\) |
| InP                                  | 3.60\(^b\) | 3.57\(^c\) | 2.5\(^c\)  |
| InSb                                 | 3.55\(^c\) | 3.53\(^c\) | 2.48\(^c\) |
| InP0.63Sb0.37/GaSb                   | 1.35\(^a\)  | 1.93\(^c\) | 2.19\(^c\) |
| InP0.69Sb0.31/InAs                   | 0.17\(^b\)  | 0.93\(^d\) | 1.63       |
| Al0.25In0.75P0.48Sb0.52/GaSb         | 0.99\(^a\)  | 1.69       | 2.09       |
| Al0.25In0.75P0.54Sb0.46/InAs         | 1.06\(^a\)  | 1.76       | 2.12       |
| Al0.25In0.75P0.58Sb0.42/InP          | 3.48\(^a\)  | 2.85       | 2.13       |
| Al0.25In0.75P0.60Sb0.35/InAs         | 3.54\(^a\)  | 2.92       | 2.16       |
| Al0.25In0.75P0.62Sb0.33/InP          | 3.84\(^a\)  | 3.26       | 2.33       |
| Al0.25In0.75P0.33Sb0.67/GaSb         | 3.08\(^b\)  | 3.29       | 2.05       |
| Al0.25In0.75P0.34Sb0.61/InAs         | 3.15\(^b\)  | 3.36       | 2.09       |
| Al0.25In0.75P0.68Sb0.32/InP          | 3.47\(^b\)  | 3.69       | 2.30       |
| Al0.25In0.75P0.69Sb0.31/InAs         | 2.72\(^b\)  | 2.78       | 1.89       |
| Al0.25In0.75P0.70Sb0.30/InP          | 2.78\(^b\)  | 2.85       | 1.94       |
| Al0.25In0.75P0.72Sb0.28/InAs         | 3.12\(^b\)  | 3.20       | 2.17       |
| AlP0.08Sb0.92/GaSb                   | 2.37\(^b\)  | 2.32       | 1.67       |
| AlP0.11Sb0.89/InAs                   | 2.44\(^b\)  | 2.39       | 1.72       |
| AlP0.30Sb0.70/InP                    | 2.79\(^b\)  | 2.75       | 1.98       |

\(^{a}\)Ref. 36, \(^{b}\)Ref. 39, \(^{c}\)Ref. 1, \(^{d}\)Ref. 40, \(^{e}\)Ref. 41.
The transverse effective charge has the same behavior of the polarity and their values are larger when the alloy is lattice-matched to InP compared to GaSb and InAs substrates.

The calculated elastic constants $C_{11}$, $C_{12}$, and $C_{44}$ of $\text{Al}_x\text{In}_{1-x}\text{P}_y\text{Sb}_{1-y}$, for some different values of Al concentration and all substrates are listed in Table III and displayed in Fig. 2. From the results, it is seen that all the elastic constants are slightly increased when Al content is enhanced whatever the used substrates. The trends of the elastic constant of the alloy when using GaSb, InAs and InP substrates are qualitatively consistent with each other. Further quantitative analysis shows that the largest elastic constant is obtained when alloy is lattice-matched to InP substrate while the smallest elastic constant is obtained when the alloy is lattice-matched to GaSb in the entire composition range. This is because the polarity of InP is higher than that of InAs and GaSb. Also, InP has a smaller nearest-neighbor distance than GaSb and InAs. The elastic constants $C_{11}$, $C_{12}$, and $C_{44}$ are inversely proportional to the nearest-neighbor distance and are directly proportional to polarity. For all substrates, the values of $C_{11}$ are larger than $C_{12}$ and $C_{44}$ values. Our results are compared with the available published data and showed good agreement. In Table III, the values of the elastic factors fulfill the stability conditions $C_{11} > 0$, $C_{44} > 0$, $C_{11} - C_{12} > 0$, $C_{11} + 2C_{12} > 0$ which shows that $\text{Al}_x\text{In}_{1-x}\text{P}_y\text{Sb}_{1-y}$ alloys are mechanically stable in their structure.

In Table IV, we show our results for the bulk $B_u$, shear $C_s$, and Young $Y_o$ moduli in $\text{Al}_x\text{In}_{1-x}\text{P}_y\text{Sb}_{1-y}$ lattice-matched to GaSb, InAs, InP substrates for some compositions ($x$). The trends of the elastic parameters with enhanced Al content in $\text{Al}_x\text{In}_{1-x}\text{P}_y\text{Sb}_{1-y}$ are qualitatively similar for all used substrates. The bulk, Young’s and shear moduli have the same behavior for the elastic constants. This is because they depend on the elastic constants $C_{11}$, $C_{12}$, and $C_{44}$. Their values are slightly increased with increasing $x$; however, the real values for InP substrate are larger than those for other substrates. The variation of $B_u$, $C_s$, and $Y_o$ moduli versus $x$ are displayed in Fig. 3. According to Pugh’s correlation, the $\text{Al}_x\text{In}_{1-x}\text{P}_y\text{Sb}_{1-y}$ alloys are ductile materials. This is because the calculated bulk/shear values for the alloys under investigation are greater than 1.75.

The bond-stretching ($\alpha$), bond-bending ($\beta$) force constants, and internal strain parameters ($\zeta$) for some compositional alloys and the different substrates are inserted in Table V and displayed in Fig. 4. Our results in this table are compared with the existing data in the literature and a good agreement is observed. From Fig. 4, we show that ($\alpha$) and ($\beta$) are slightly increased with increasing Al concentration; however, $\zeta$ has the smallest values and is not affected by enhancing the compositional parameter whatever the used substrates.

The LO phonon frequency’s Debye temperature is a significant parameter in the research of polaron mobility at
finite temperatures. This temperature is given by $\theta_D = \frac{\hbar}{k}$, where $k$ is the Boltzmann’s constant. The Debye temperature ($\theta_D$) for the Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ alloy for different substrates (GaSb, InAs, and InP) as a function of composition ($x$) is displayed in Fig. 5. It is seen that the Debye temperature is increased nonlinearly by increasing composition. Also, the Debye temperature for the Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ alloy
lattice-matched to the InP substrate has higher values than the InAs and GaSb substrates.

The LO and TO phonon frequencies are determined by the Lyddane–Sachs–Teller relation.34,35 The calculated LO(Γ) and TO(Γ) phonon frequencies and their dependence on composition for Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ alloy at different substrates are shown in Fig. 6. As observed in our results, the frequencies of these two modes increases non-linearly

### Table IV Elastic moduli of AIP, AlSb, InP, InSb and their alloys lattice-matched to different substrates for some compositions (x) at room temperature and normal pressure

| Compound | B$_u$ (10$^{11}$ dyn/cm$^2$) | Error % | Error % | Error % |
|----------|-------------------------------|---------|---------|---------|
| AIP      | 8.23                          | 9.70    | 3.72    |
|          | 8.25$^b$                      | 0.2     | 11.2$^b$| 4.42$^b$| 15.8   |
|          | 8.24$^c$                      | 0.1     |         |         |        |
| AlSb     | 5.54                          | 6.62    | 2.54    |
|          | 5.80$^a$                      | 4.48    | 5.90$^a$| 12.2    | 2.20$^a$| 15.4   |
| InP      | 6.16                          | 7.29    | 2.79    |
|          | 7.23$^a$                      | 14.7    | 6.10$^a$| 19.5    | 2.25$^a$| 24     |
| InSb     | 4.05                          | 4.8     | 1.85    |
|          | 4.56$^a$                      | 11.1    | 4.15$^a$| 15.6    | 1.54$^a$| 20.1   |
|          | 3.89$^d$                      | 4.1     |         |         |        |
| InP$_{0.6283}$Sb$_{0.3717}$/GaSb | 5.23 | 6.2 | 2.38 |
| InP$_{0.6899}$Sb$_{0.3101}$/InAs | 5.37 | 6.37 | 2.44 |
| Al$_{0.25}$In$_{0.75}$P$_{0.4754}$Sb$_{0.5246}$/GaSb | 5.30 | 6.30 | 2.41 |
| Al$_{0.24}$In$_{0.76}$P$_{0.5355}$Sb$_{0.4645}$/InAs | 5.44 | 6.45 | 2.47 |
| Al$_{0.22}$In$_{0.78}$P$_{0.8380}$Sb$_{0.1620}$/InP | 6.17 | 7.31 | 2.80 |
| Al$_{0.35}$In$_{0.65}$P$_{0.3299}$Sb$_{0.6701}$/GaSb | 5.41 | 6.43 | 2.47 |
| Al$_{0.35}$In$_{0.65}$P$_{0.3886}$Sb$_{0.6114}$/InAs | 5.54 | 6.59 | 2.53 |
| Al$_{0.35}$In$_{0.65}$P$_{0.6638}$Sb$_{0.3362}$/InP | 6.25 | 7.41 | 2.84 |
| Al$_{0.35}$In$_{0.65}$P$_{0.2485}$Sb$_{0.7515}$/GaSb | 5.55 | 6.60 | 2.53 |
| Al$_{0.35}$In$_{0.65}$P$_{0.2485}$Sb$_{0.7515}$/InAs | 5.68 | 6.76 | 2.59 |
| Al$_{0.35}$In$_{0.65}$P$_{0.2485}$Sb$_{0.7515}$/InP | 6.38 | 7.57 | 2.90 |
| AlP$_{0.0589}$Sb$_{0.9411}$/GaSb | 5.69 | 6.78 | 2.60 |
| AlP$_{0.1149}$Sb$_{0.8851}$/InAs | 5.82 | 6.94 | 2.66 |
| AlP$_{0.3966}$Sb$_{0.6034}$/InP | 6.54 | 7.78 | 2.98 |

$^a$Ref. 36, $^b$Ref. 48, $^c$Ref. 21, $^d$Ref. 50.

Fig. 3 Elastic moduli (Y, C$_s$, B$_u$) of Al$_x$In$_{1-x}$P$_y$Sb$_{1-y}$ for GaSb, InAs and InP substrates as a function of composition (x).
with increasing composition from 0 to 1. Subsequently, the transverse effective charge is increased with increasing composition, since the transverse effective charge increases with the composition. The stronger the bond level, the higher the vibration frequency, and vice versa. The optical phonon mode, which is dependent on the bond level, has the maximum vibrational frequency at the Γ-point. The phonon frequencies ($\omega_{LO}, \omega_{TO}$) in $10^{13}$ s$^{-1}$ of the central optical phonon modes for AlP, AlSb, InP, and InSb compounds were calculated to be (11.25, 9.98),
(8.87, 8.53), (6.81, 6.19), and (3.22, 3.03), respectively. It is noted that the calculated values of \( \omega_{\text{LO}} \) and \( \omega_{\text{TO}} \) are in reasonable agreement with the experimental and published data\(^{22,44,53,54} \) justifying the reliability of our calculated data. The obtained data for the calculated longitudinal and transverse phonon frequencies of Al\(_{x}\)In\(_{1-x}\)P\(_{y}\)Sb\(_{1-y}\) at different values of composition may be taken as a reference for future experimental work. It is observed that the Al\(_{x}\)In\(_{1-x}\)P\(_{y}\)Sb\(_{1-y}\) alloy has higher values of the phonon frequencies \( \omega_{\text{LO}} \) and \( \omega_{\text{TO}} \) for the InP substrate than the other two substrates (GaSb and InAs).

It is well recognized that studying transport and optical properties in polar semiconductors necessitates an understanding of the electron-LO lattice vibration coupling, which cannot be overlooked. The well-known Fröhlich coupling constant is a measure of the interaction between electrons

![Graph](image1)

**Fig. 5** Debye temperature (\( \theta_D \)) of Al\(_{x}\)In\(_{1-x}\)P\(_{y}\)Sb\(_{1-y}\) for GaSb, InAs and InP substrates as a function of composition (x).

![Graph](image2)

**Fig. 6** Phonon frequencies (\( \omega_{\text{LO}} \) and \( \omega_{\text{TO}} \)) of Al\(_{x}\)In\(_{1-x}\)P\(_{y}\)Sb\(_{1-y}\) for GaSb, InAs, and InP substrates as a function of composition (x).

![Graph](image3)

**Fig. 7** Fröhlich coupling constant (\( \alpha_F \)) of Al\(_{x}\)In\(_{1-x}\)P\(_{y}\)Sb\(_{1-y}\) for GaSb, InAs and InP substrates as a function of composition (x).
Table VI  Sound velocity propagating along with the three major directions [100], [110] and [111] of AlP, AISb, InP, InSb and their alloys lattice-matched to different substrates for some compositions (x) (in 10^5 cm/s) at room temperature and normal pressure

| Compound | [100] | | | | [110] | | | | | [111] | | |
|----------|------|------|------|------|------|------|------|------|------|------|------|------|
|          | $V_{LA}$ | Error % | $V_{TA1-TA2}$ | Error % | $V_{LA}$ | Error % | $V_{TA1}$ | Error % | $V_{TA2}$ | Error % | $V_{LA}$ | Error % | $V_{TA1-TA2}$ | Error % |
| AlP      | 7.40  | 4.74  | 7.91  | 3.97  | 7.40  | 3.97  | 4.74  | 8.05  | 4.24  | 8.05  | 4.24  |  |  |
|          | 7.97a | 7.1   | 5.09a | 6.8   | 8.44a | 6.2   | 4.26a | 6.8   | 5.09a | 6.8   | 8.59a | 6.2   | 4.56a | 7     |
| AISb     | 4.57  | 2.90  | 4.83  | 2.43  | 4.57  | 2.90  | 4.83  | 2.43  | 4.57  | 2.90  | 4.83  | 2.43  | 4.92  | 2.60  |
|          | 4.53a | 0.8   | 3.09a | 6.1   | 4.99a | 3.2   | 2.28a | 6.5   | 3.09a | 6.1   | 5.13a | 4     | 2.57a | 1.1   |
| InP      | 4.54  | 2.88  | 4.80  | 2.41  | 4.54  | 2.88  | 4.80  | 2.41  | 4.54  | 2.88  | 4.80  | 2.41  | 4.89  | 2.58  |
|          | 4.62a | 1.7   | 3.04a | 5.2   | 4.80a | 5.6   | 2.17a | 11    | 3.04a | 5.2   | 5.23a | 6.5   | 2.49a | 3.6   |
| InSb     | 3.36  | 2.13  | 3.55  | 1.79  | 3.36  | 2.13  | 3.55  | 1.79  | 3.36  | 2.13  | 3.55  | 1.79  | 3.62  | 1.91  |
|          | 3.38  | 0.59  | 2.29a | 6.9   | 3.74a | 5     | 1.63a | 9.8   | 2.29a | 6.9   | 3.86a | 6.2   | 1.88a | 1.5   |
| InP$_{0.628}$Sb$_{0.3717}$/GaSb | 3.99  | 2.53  | 4.23  | 2.12  | 3.99  | 2.53  | 4.23  | 2.12  | 3.99  | 2.53  | 4.30  | 2.27  |  |  |
| InP$_{0.6899}$Sb$_{0.3101}$/InAs | 4.07  | 2.58  | 4.31  | 2.16  | 4.07  | 2.58  | 4.31  | 2.16  | 4.07  | 2.58  | 4.39  | 2.31  |  |  |
| Al$_{0.25}$In$_{0.75}$P$_{0.4754}$Sb$_{0.5246}$/GaSb | 4.11  | 2.61  | 4.36  | 2.19  | 4.11  | 2.61  | 4.36  | 2.19  | 4.11  | 2.61  | 4.43  | 2.34  |  |  |
| Al$_{0.25}$In$_{0.75}$P$_{0.5316}$Sb$_{0.4684}$/InAs | 4.20  | 2.66  | 4.44  | 2.23  | 4.20  | 2.66  | 4.44  | 2.23  | 4.20  | 2.66  | 4.52  | 2.38  |  |  |
| Al$_{0.25}$In$_{0.75}$P$_{0.8380}$Sb$_{0.3620}$/InP | 4.66  | 2.96  | 4.94  | 2.48  | 4.66  | 2.96  | 4.94  | 2.48  | 4.66  | 2.96  | 5.02  | 2.65  |  |  |
| Al$_{0.37}$In$_{0.63}$P$_{0.3299}$Sb$_{0.6701}$/GaSb | 4.27  | 2.71  | 4.52  | 2.27  | 4.27  | 2.71  | 4.52  | 2.27  | 4.27  | 2.71  | 4.60  | 2.43  |  |  |
| Al$_{0.37}$In$_{0.63}$P$_{0.3886}$Sb$_{0.6114}$/InAs | 4.35  | 2.76  | 4.61  | 2.32  | 4.35  | 2.76  | 4.61  | 2.32  | 4.35  | 2.76  | 4.69  | 2.47  |  |  |
| Al$_{0.43}$In$_{0.57}$P$_{0.0683}$Sb$_{0.9317}$/InP | 4.83  | 3.07  | 5.12  | 2.57  | 4.83  | 3.07  | 5.12  | 2.57  | 4.83  | 3.07  | 5.21  | 2.75  |  |  |
| Al$_{0.75}$In$_{0.25}$P$_{0.1919}$Sb$_{0.8081}$/GaSb | 5.06  | 3.21  | 5.35  | 2.69  | 5.06  | 3.21  | 5.35  | 2.69  | 5.06  | 3.21  | 5.45  | 2.87  |  |  |
| Al$_{0.75}$In$_{0.25}$P$_{0.5368}$Sb$_{0.4632}$/InP | 5.01  | 3.18  | 5.34  | 2.67  | 5.01  | 3.18  | 5.34  | 2.67  | 5.01  | 3.18  | 5.36  | 2.69  |  |  |
| Al$_{0.0598}$Sb$_{0.9402}$/GaSb | 4.66  | 2.96  | 4.94  | 2.49  | 4.66  | 2.96  | 4.94  | 2.49  | 4.66  | 2.96  | 5.02  | 2.65  |  |  |
| Al$_{0.0899}$Sb$_{0.9101}$/InAs | 4.76  | 3.02  | 5.04  | 2.54  | 4.76  | 3.02  | 5.04  | 2.54  | 4.76  | 3.02  | 5.13  | 2.71  |  |  |
| Al$_{0.3966}$Sb$_{0.6034}$/InP | 5.33  | 3.38  | 5.64  | 2.84  | 5.33  | 3.38  | 5.64  | 2.84  | 5.33  | 3.38  | 5.74  | 3.03  |  |  |

aRef. 1.
and LO phonons which is given by
\[
\alpha_F = 0.5 \left( \frac{1}{r^c} - 1 \right) \frac{e^2}{\hbar} \frac{(h/2m*)^{1/2}}{m^*}
\]
where \(m^*\) is the electron effective mass. It is highly influenced by the crystal's ionic polarization, which is linked to the dielectric constants. Figure 7 shows Fröhlich coupling constant (\(\alpha_F\)) of \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) for GaSb, InAs and InP substrates as a function of composition, which shows that it increases with increasing composition from 0 to 0.25 and decreases from 0.25 to 1 for two substrates (GaSb and InAs). It is noted that this constant increases by increasing composition from 0 to 0.4 and decreases from 0.4 to 1 for the InP substrate. The constant has higher values for the InP substrate than the other two substrates.

The longitudinal and transverse sound velocity propagating along with the three major directions [100], [110], and [111] for the \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) alloy for different values of composition and different substrates are listed in Table VI and displayed in Fig. 8a, b, and c. The calculated data of the sound velocity (in \(10^5\) cm/s) along the three major directions for the considered alloys are in good agreement with the available results (in Table VI). The calculated results of the sound velocity at the rest values of composition may be taken as a reference for future experimental work. It can be observed from Fig. 8 that the longitudinal and transverse sound velocity for the three major directions [100], [110], [111] of \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) is increased with increasing composition. It is noted that the acoustic velocity for \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) lattice-matched to InP substrate has higher values than that for InAs and GaSb substrates.

**Conclusion**

Based on EPM within the virtual crystal approximation (VCA), the energy band gaps of \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) lattice-matched to GaSb, InAs, and InP substrates and its constituents are calculated for different Al concentrations. The polarities and transverse effective charge of the \(\text{Al}_{x}\text{In}_{1-x}\text{P}_{y}\text{Sb}_{1-y}\) system are determined from the symmetric and anti-symmetric form factors at \(G(1,1,1)\) using Vogl’s definition. The elastic constants \(C_{11}, C_{12}, C_{44}\), and their related elastic moduli \(B_0, C_1\) and \(Y_0\) are calculated and their variation with Al concentration has been examined. Also, the bond-stretching (\(\alpha\)), bond-bending (\(\beta\)) force constants, and internal strain parameters (\(\zeta\)) are calculated for the used substrates. The
Debye temperature and the Fröhlich coupling constant have been determined at different values of composition lattice-matched to different substrates. The phonon frequencies and the sound velocity for different substrates at various compositions have been studied. Most of our results are compared with the known data in the literature and showed good agreement. Our data cover a wide range accessed by varying compositions and substrates. The calculated quantities in this work may be of great interest for both theoretical experimental research and devise applications; it opens up the possibility of developing new electronic devices.

Acknowledgments The authors thank the Science, Technology & Innovation Funding Authority (STDF) in cooperation with The Egyptian Knowledge Bank (EKB).

Funding Open access funding provided by The Science, Technology & Innovation Funding Authority (STDF) in cooperation with The Egyptian Knowledge Bank (EKB).

Conflict of interest The authors declare that they have no conflict of interest.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons licence, and indicate if changes were made. The images or other third party material in this article are included in the article’s Creative Commons licence, unless indicated otherwise in a credit line to the material. If material is not included in the article’s Creative Commons licence and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this licence, visit http://creativecommons.org/licenses/by/4.0/.

References

1. I. Vurgaftman, J. áR Meyer, and L. áR Ram-Mohan, J. Appl. Phys. 89, 5815 (2001).
2. A.R. Degheidy, A.M. AbuAli, and E.B. Elkenany, Appl. Phys. A 127, 1 (2021).
3. M. Henini and M. Razeghi, Optoelectronic Devices: III Nitrides (Elsevier, 2004).
4. A.R. Degheidy and E.B. Elkenany, Chinese Phys. B 26, 086103 (2017).
5. A.R. Degheidy, and E.B. Elkenany, Chin. Phys. B 24, 94302 (2015).
6. I.F. Al Maaitah and E.B. Elkenany, Comput. Condens. Matter e00640 (2022).
7. E.B. Elkenany, A.R. Degheidy, and O.A. Alfrnwani, SILICON 11, 919 (2019).
8. A.R. Degheidy, E.B. Elkenany, and O.A. Alfrnwani, Comput. Condens. Matter 15, (2018).
9. A.R. Degheidy, E.B. Elkenany, and O.A. Alfrnwani, Comput. Condens. Matter 16, e00300 (2018).
10. A.R. Degheidy, E.B. Elkenany, and O.A. Alfrnwani, Silicon 9, 183 (2017).
11. E.B. Elkenany, Silicon 8, (2016).
12. A.R. Degheidy and E.B. Elkenany, Phys. B Condens. Matter 456, (2015).
13. A.R. Degheidy, E.B. Elkenany, M.A.K. Madkour, and A.M. AbuAli, Comput. Condens. Matter 16, e00308 (2018).
14. A. R. Degheidy and E. B. Elkenany, Semiconductors 47, 1283 (2013).
15. M. Othman, E. Kasap, and N. Korozlu, J. Alloys Compd. 496, 226 (2010).
16. M. Othman, S. Salih, M. Sedighi, and E. Kasap, Results Phys. 14, 102400 (2019).
17. M.S. Othman, Al-Mustansiriyyah J. Sci. 31, 120 (2020).
18. B. Ghebouli, M.A. Ghebouli, and M. Fatmi, Phys. B Condens. Matter 406, 2521 (2011).
19. B. Ghebouli, M.A. Ghebouli, M. Fatmi, and N. Lebgaa, Mater. Chem. Phys. 128, 195 (2011).
20. M.A. Ghebouli, H. Choutri, N. Bouarissa, and B. Ghebouli, J. Solid State Chem. 192, 161 (2012).
21. A. Bouhemadou, M.A. Ghebouli, B. Ghebouli, M. Fatmi, S. Bin-Omran, E. Ucgun, and H.Y. Ocak, Mater. Sci. Semicond. Process. 16, 718 (2013).
22. E.B. Elkenany, Infrared Phys. Technol. 103720 (2021).
23. A. R. Degheidy and E. B. Elkenany, Thin Solid Films 539, 365 (2013).
24. A. R. Degheidy, A. M. AbuAli, and E. B. Elkenany, (2022).
25. A. R. Degheidy and E. B. Elkenany, Mater. Chem. Phys. 157, (2015).
26. A.R. Degheidy and E.B. Elkenany, J. Alloys Compd. 652, 379 (2015).
27. H.Y. Fan, Phys. Rev. 82, 900 (1951).
28. C. Keffer, T.M. Hayes, and A. Bienenstock, Phys. Rev. Lett. 21, 1676 (1968).
29. C. Keffer, T.M. Hayes, and A. Bienenstock, Phys. Rev. B 2, 1666 (1970).
30. E.B. Elkenany, Phys. Scr. 96, 105701 (2021).
31. E.B. Elkenany and M.S.H. Othman, Phys. Scr. 96, 125718 (2021).
32. P. Harrison, Quantum Wells, Wires and Dots (Wiley Online Library, 2016).
33. L. Vegard, Zeitschrift Für Phys. 5, 17 (1921).
34. S.Y. Davydov and S.K. Tikhonov, Semiconductors 32, 947 (1998).
35. C. Kittel and P. McEuen, Introduction to Solid State Physics (New York: Wiley, 1976).
36. S. Adachi, Properties of Group-Iv, III-v and II-VI Semiconductors (John Wiley & Sons, 2005).
37. H.Y. Wang, J. Cao, X.Y. Huang, and J.M. Huang, ArXiv Prepr. ArXiv1204.6102 (2012).
38. K.C. Pandey and J.C. Phillips, Phys. Rev. B 9, 1552 (1974).
39. S. Tiwari and D.J. Frank, Superlattices Microstruct. 20, 265 (2005).
40. S. Adachi, J. Phys. C Solid State Phys. 20, 14, 6102 (1987).
41. A. Bechiri, F. Bennakhlefou, and N. Bouarissa, Mater. Chem. Phys. 77, 507 (2003).
42. S. Zerroug, F.A. Sahraoui, and N. Bouarissa, Mater. Lett. 60, 546 (2006).
43. W.A. Harrison and S. Ciraci, Phys. Rev. B 10, 1516 (1974).
44. N. Bouarissa, S. Bougouffl and A. Kamli, Semicond. Sci. Technol. 20, 265 (2005).
45. P. Vogl, J. Phys C Solid State Phys. 11, 251 (1978).
46. X. Zhang, C. Ying, Z. Li, and G. Shi, Superlattices Microstruct. 52, 459 (2012).
47. M. Krijn, Semicond. Sci. Technol. 6, 27 (1991).
48. N. Bouarissa, Mater. Sci. Eng. B 100, 280 (2003).
49. S.F. Pugh, London Edinburgh Dublin Philos. Mag. J. Sci. 45, 823 (1954).
50. S. Lukel, F. Okbi, and H. Meradji, Optik (Stuttg). 127, 3755 (2016).
51. S. de Gironcoli, S. Baroni, and R. Resta, Phys. Rev. Lett. 62, 2853 (1989).
52. H. Baaziz, Z. Charifi, and N. Bouarissa, Mater. Chem. Phys. 68, 197 (2001).
53. C. Kittel, P. McEuen, and P. McEuen, Introduction to Solid State Physics (New York: Wiley, 1996).
54. S. Ehsanfar, F. Kanjouri, H. Tashakori, and A. Esmailian, J. Electron. Mater. 46, 6214 (2017).
55. J.T. Devreese, Polarons in Ionic Crystals and Polar Semiconductors: Antwerp Advanced Study Institute 1971 on Fröhlich Polarons and Electron-Phonon Interaction in Polar Semiconductors (North-Holland Publishing Company, 1972).
56. S. Adachi, J. Appl. Phys. 58, R1 (1985).

Publisher's Note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.