Polyvinyl Alcohol-Cellulose Acetate Composite Reverses Osmosis Membranes: I. Synthesis and Characterization

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Abstract

To overcome the high salinity water resources problem, reverse osmosis (RO) process is considered one of the most required techniques due to its high portable water quality produced. Optimization of membranes quality and function is still pursued and considered as of an important added value to the RO processes. In the present study, thin film PVA/CA+PEG membranes were prepared and the PVA layer was crosslinked by a varying maleic acid concentration at different reaction periods. The polymer composite is to be employed for the reverse osmosis process of brackish, saline and sea water purification. Optimization of reaction conditions and parameters such as salt rejection and water flux that affect on the membrane efficiency were verified. Structural and functional characterization of the PVA/CA composite membranes through Fourier Transform Infrared (FT-IR), thermogravimetric analysis (TGA), X-ray diffraction pattern, scanning electron microscopy (SEM), swelling behavior and membrane mechanical properties were monitored.

Keywords: Reverse osmosis membrane; Desalination; Composite film; Polyvinyl alcohol; Cellulose acetate; Polyethylene glycol; Crosslinking agents; Salt rejection.

Introduction

Membrane and membrane processes were first introduced as an analytical tool in chemical and biomedical laboratories that developed very rapidly into industrial products and methods with significant technical and commercial importance for the separation of many inorganic and organic entities [1-4]. In order to achieve high values of permeability and selectivity, the active layer should be ultrathin and hydrophilic [5-9]. Reverse osmosis membranes are prepared by different techniques, amongst micro porous polymeric membrane, which finds many applications in the fields of dehydration of organic mixtures, water treatment and organic-organic separation [10,11]. A most successful process known as pre-evaporation, the dehydration of aqueous-organic mixtures, well established for the dehydration of isopropanol, acetone, tetrahydrofuran, ethylene glycol and methylene chloride [12-14].

PVA membranes are widely used for dehydration of organics [15-18] and giving good flux but with a poor selectivity due to the negative impact of membrane swelling. To enhance the membrane performance, a variety of membrane modification techniques are employed such as crosslinking, blending and grafting of several chemical agents to the original polymer matrix [19,20]. Among those chemical treatments of membranes, crosslinking has been used extensively to modify membranes physicochemical properties, such as crystallinity, hydrophobicity, and mechanical strength [22-24]. PVA is recognized by its high hydrophilic character, good film-forming properties and outstanding physical and chemical stability, it is then considered as an excellent membrane material for preparation of hydrophilic UF and MF membranes [25].

Materials and Methods

Chemicals

Polyvinyl alcohol (PVA) of a molecular weight 133,000 g/mol was supplied by Fisher Scientific. Cellulose acetate (CA) of 39.6% acetyl content average (Mw = 50,000), polyethylene glycol (MW avg. 400), maleic acid and methanol were provided from Aldrich co. Deionized water (DI) was obtained from a Millipore Milli-Q filtration system with a resistivity of 18 cm. solvents and inorganic salts were of reagent grade and used directly without further purification. Maleic acid and methanol were provided from Aldrich co. Deionized water (DI) was obtained from a Millipore Milli-Q filtration system with a resistivity of 18 cm. solvents and inorganic salts were of reagent grade and used directly without further purification.

Preparation of microporous Poly (vinyl alcohol) support membrane: 1-Poly (vinyl alcohol) casting solution was prepared by dissolving 5wt % (PVA) in water at 90°C with constant stirring until homogenous solution was obtained.

2- 0.001 wt % of the maleic acid as crosslinking agent was then added.

3-The polymer solution was then allowed to cast over a glass plate and subsequently placed in an oven at 50°C overnight and finally washed with deionized water before further treatment. The effect of maleic acid concentration on the support membrane (PVA) durability was followed through varying its concentration from 0.001 to 0.6 wt % at reaction periods of 10 to 90 minutes. Figure 2 illustrates the reaction mechanism between PVA and fumaric acid as an analog to that of PVA and maleic acid.

Fabrication of composite membranes (dip coating method): Cellulose acetate was dissolved in methanol at different concentrations of 5, 25, 40, and 60 wt %. The solution was deposited coherently onto the top surfaces of different samples of the PVA support membrane.

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PEG was then added to the casting mixture to increase the porosity of the cast membrane while maintaining a high polymer content (15-20%) and, therefore, sufficiently high viscosity of the casting mixture [26,27]. Graphical representation of the final polymer composite membrane structure after the casting process is shown in figure 2.

The composite membrane environmental location is schematically represented in figure 3.

Membrane characterization and functional features: Structural and functional studies of the PVA/CA grafted RO membranes were conducted and evaluated through the following analytical, functional and performance techniques.

FT-IR structural study: Analysis has been carried out using Infrared spectroscopy, Genisis Unicam FT-IR Spectrophotometer. The dimensions of the window, to which the film was fixed, were 1.5 X 3 cm. The polymer membrane was finally chopped and KBr pellet was prepared. The FT-IR spectrum was investigated in the frequency range 4000-500 cm⁻¹. FT-IR studies were conducted at the Central Laboratory of Ain-Shams University- Abbassia, Cairo.

Surface and topographical studies of membrane: The surface morphologies of the hydro gels as well as the surface of the membranes were investigated by SEM images using a JSM-5800 instrument, JEOL, (Japan). The polymer films were gold coated before the study. Photographs have been taken X100 magnification. Tapping mode on a commercial AFM (Picoscan, Molecular Imaging, Tempe, AZ, USA) was also employed to aid with topography and phase images of the polymer membrane surfaces. SEM and AFM images were performed at National Center for Radiation Research and Technology, Atomic energy Authority, Egypt.

Mechanical properties: Mechanical measurements were carried out through measuring both tensile strength and elongation percent by using an Instron (Model -1195, England), National Center for Radiation Research and Technology, Atomic energy Authority, Egypt.

Thermal stability profile (TGA): Thermal stability and solid state dehydration changes of membrane samples were carried out using a Shimadzu DT-60H thermal analyzer, Shimadzu, Kyoto, Japan, the measurements were carried out at 500°C at a heating rate of 15° / minute, National Center for Radiation Research and Technology, Atomic energy Authority, Egypt.

G-Reverse Osmosis functional measurements: DDS Reverse
osmosis system available at National Center for Radiation Research and Technology, Atomic energy Authority, Egypt (Figure 4).

Results and Discussion

PVA, CA, PVA/CA composite membrane structural characterization

FT-IR structural features: PVA spectral features are well recognized and as shown in figure 5a, where the –OH familiar groups are exhibited at the absorption range of 3000-3500 cm\(^{-1}\). When the characteristic –OH groups of the PVA interchain with the acyl groups of maleic acid, where the stretching vibration at 1647.5 cm\(^{-1}\) is shifted to 1601.3 cm\(^{-1}\) with the prevalence of the -OH groups and the aliphatic nature of the PVA and the –CO of the maleic acid in the regions of 1500-1800, 1200-1400cm\(^{-1}\) and 1100cm\(^{-1}\) where crosslinking effect is affecting not only peaks intensity but also shifting of locations. However, new covalence can be delicately observed but indication of crosslinking action can only be cited at 2871.6 cm\(^{-1}\) and as exhibited when comparing figure 5a and 5b.

CA membrane FT-IR spectrum is represented in figure 5c. Vibration bands at 3480 cm\(^{-1}\), 3043 cm\(^{-1}\) and 1754 cm\(^{-1}\) are exhibited as prominent for the –OH, C-H stretching and C=O vibration of ester group. Stretching bands of the alkane groups (CH\(_3\) and CH\(_2\)) are shown in the range 1373-1519 cm\(^{-1}\) while bands at 1051-1253 cm\(^{-1}\) correspond to stretching modes of C-O single bond of esters and ethers.

The above FT-IR absorption spectrum of native PVA, crosslinked by maleic acid and of added CA to the mixture indicate on covalency occurrence between the PVA and CA layers which in turn needed to be mutually adhered for durability. Addition of PEG onto the PVA/CA maleic acid composite was thought for membrane durability; FT-IR spectrum of which is exhibited in figure 5d. The spectrum exhibits polyethylene glycol absorptions of its primary alcohol groups observed in the region 3378 cm\(^{-1}\). Hence these absorptions, which comprise stretching and bending vibrations restricted to C-C stretch, C-O stretch, -CH stretch (methylene absorptions) and the C-H bending. The -OH stretching vibration is also observed in the region 3378 cm\(^{-1}\) exhibiting hydrogen bonded nature. The strong bands shown at 966 cm\(^{-1}\) and 849 cm\(^{-1}\) are indicative to the C-C stretching.

The above FT-IR structural features of the native PVA, maleic acid added, CA layered and PEG final addition indicate a composite formed polymer mixture membrane durability of which and functionality will be justified when tested for the purpose of RO desalination.

Scanning electron microscopy (SEM): Revealing the microstructure of the various components of the membrane could be

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**Figure 5:** FT-IR of synthetic composite membrane films.

**Figure 6:** SEM images of pure PVA at different maleic acid concentration.
regarded as a useful tool when interpreting durability and function values. Accordingly, figure 6 exhibits the scanning electron micrograph image for PVA-cellulose, PVA/cellulose/PEG materials at different maleic acid concentrations. The SEM images revealed presence of cellulose fiber entanglements on the surface of the membranes. There are slight changes in the surface morphology by addition of maleic acid an increasing porosity due to the increase of the number of pores on the surface. This result evidences that the addition of maleic acid to PVA will increase the performance and efficiency of composite membranes. Mechanical properties are known to be strongly dependent on their morphologies [28], the homogenous morphology could have influenced the tensile properties of the blends obtained in case of PVA & maleic acid in which the homogenous morphology is accompanied by possessing high tensile and elongation

Figure 7 (a,b,c) illustrates the SEM images of cellulose membrane before soaking with PVA and after PVA soaking respectively. Thickening of cellulose fibers, figure 8a, are shown to occur after being coated with PVA, figure 8b and the inner layers become less visible. Although, some diffused features are visible in the uncoated cellulose membranes due to the presence of some open fibrils, but the fibers are apparently smoothened due to coating of the open fibrils with PVA contrary to PVA membranes, figure 7, where smooth and featureless surface without pores are noted. Upon crosslinking by PEG to the CA/ PVA material the morphological feature seems to be more compact and as observed from figure 8c. As acetone was used during membrane casting of the CA, it is expected that the molecules are to be immobilized in a glassy state and no pores were formed [29].

**Tapping mode atomic force microscopy (AFM):** In order to investigate clearly the surface morphology of the composite membranes, atomic force microscopy (AFM) was employed, results of which are illustrated in figure 8 which show the images of two and three dimensional scans (scale 8 µm) of the CA, PVA and PVA+CA&PEG membranes [30]. It could be seen from the images, the membrane without PEG showed nearly a unique and characteristic ridge-and-valley structure consistent [31].

The images demonstrate clearly homogenous morphology and distribution features of CA and PVA, while the PVA+CA & PEG membrane show many irregularities more than that of PVA, with new protuberances and unevenly distributed features. These irregularities may be due to the insufficient coverage of PEG on membrane surface. As a result, the membrane surface was not completely covered by PEG and the uncovered area showed lower position. Therefore, the roughness parameters of the membrane with PEG increased compared to the membrane without its presence [32].

**Thermogravimetric analysis (TGA):** The thermal gravimetric analysis (TGA) was performed on the selected polymers evaluated in this study in order to examine the influence of their structural differences on their degradation behavior [33]. Investigation of thermal gravimetric analysis (TGA) of the prepared membranes is presented in figure 9 and is interpreted as follow:

According to figure 9A, i.e. PVA thermal degradation, it is seen that the degradation of all synthetic membranes occurs in three steps:

- The range of first step is from room temperature to 330°C in case of PVA/CA membrane. This step represents the evolution of the volatile matter and/or the evaporation of residual absorbed water.

- The second step starts from 330 to 380°C which represents the main thermal degradation of the CA chains.

- The third step which takes place over 381°C for PVA/CA membrane which symbolizes the carbonization of the degraded products to ash.

Figure 9B, illustrates the degradation of PVA/CA/PEG which takes place also in three steps but is emphasized by more thermal stability than the second composition due may be to addition of PEG to the composition, which the thermal degradation from 402°C.

Such study proves that the crosslinked PVA/CA membrane by
In case of PVA, indicated that; hot water is the suitable solvent for the preparation of PVA membranes that are homogenous and have good properties.

**Effect of cross-linking agent addition:** The pervaporation performance of the PVA membrane can be affected by the nature of cross-linking agent used with different dicarboxylic acids, as cross-linking agents. Maleic acid was selected as the PVA's cross-linking agent for the preparation of PVA/CA composite membrane [41]. Accordingly the effect of maleic acid at 0.001 wt % on the membrane swelling and gelation parameters were studied, results of which are recorded in table 4.

**Effect of reaction temperature and time:** It is important to note that the reaction temperature should not exceed 120°C (the temperature of membrane formation) [42,43]; otherwise it may affect the polymer deformation, water dehydration. The effect of reaction temperature increase on the swelling ability is illustrated in figure 10.

The composite membrane preparation of PVA/CA with maleic acid as cross-linking was carried out at different reaction times in the range of 45-120 minutes, figure 11. As can be seen as the reaction time of the membrane preparation is increased, both conversion and water insoluble part % increase, leading to the formation of condensed network structure that could prevent the penetration of water molecules, and in turn decrease the swelling %.

**Effect of membrane thickness:** As shown in table 5, by increasing the membrane thickness, the salt rejection increases while the water flux decreases. This suggests a blocking effect that hinders, to some extent, the movement of water molecules through the membrane [44].

**Effect of chemical treatment:** Alkaline treatment of the PVA+CA&PEG membranes (impregnation in Na₂CO₃ at room temperature), was carried out to confer ionic character which may result in improving the hydrophilic properties of the grafted membranes. The effect of alkali membrane treatment on the membrane swelling and gelation parameters is offered in table 6. Alkaline treatment of PEG results in converting the free-COOH groups into the strong electrolyte -COO⁻Na⁺ which possesses higher swelling behaviour due to the broken hydrogen bonding taking place by such chemical conversion of –COOH into its acrylate (-COO-NA⁺).

**Effect of solvent evaporation:** In Tables 7, 8 in contrast, low evaporation temperature results in a membrane of high solvent content and the possibility of existence of large pores on the membrane becomes high, this leads to high water flux with low salt rejection [45]. However, the increase in evaporation time at fixed temperature leads to a decrease in water flux with increasing salt rejection.

**Effect of feed concentration:** In case of salt rejection, the decrease of salt rejection by increasing the feed concentration can be explained by, the passage of salt through the membrane in terms of several

### Table 1: The mechanical properties for some synthetic composite RO membranes.

| Specimen | Yield MPa | Yield % | Break MPa | Break % |
|----------|-----------|---------|-----------|---------|
| 1        | 34.62     | 11.60   | 13.46     | 34.60   |
| 2        | 38.75     | 18.00   | 58.5      | 49.88   |
| 3        | 63.5      | 21.38   | 20.88     | 137.0   |
| 4        | 64.9      | 72.4    | 75.0      | 269.6   |

Mean: 40.73, Standard Deviation: 18.33, Coefficient of Variation: 7.52

Maximum: 64.9, Minimum: 11.20, Coefficient of Variation: 7.52

### Table 2: Effect of CA addition on reverse osmosis parameters.

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### Table 4: Effect of cross-linking on swelling.

| Co-polymer | P | P/Co-P | Untreated | Treated with .001 % maleic acid |
|------------|---|--------|-----------|---------------------------------|
| CA+PVA     | PEG | 2:1    | 30        | 2.74                            |
|            |     | 40     | 18.93     | 5.38                            |

PEG has influenced its thermal stability by nearly 20°C [34,35].

Mechanical properties: In table (1) shows that the addition of PEG enhanced the resistance of CA and probably reflected the interactions between the free hydroxyl groups of PEG and the chains of CA [36], free COOH groups which enhance the formation of hydrogen bonding between the free hydroxyl groups of PEG and the chains of CA [36], and decreasing gelation % [37]. Accordingly the effect of maleic acid at 0.001 wt % on the membrane swelling and gelation parameters were studied, results of which are recorded in table 4.

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|---|--------|-----------|---------------------------------|
|   | ratio (%) | Swelling (%) | Conv (%) | Gel. (%) | Swelling (%) | Conv (%) | Gel. (%) |
| CA+PVA | PEG | 2:1 | 20 | 49.84 | 81.75 | 68.29 | 110 | 61 | 59.16 |
|       |     | 4:1 | 40 | 52.64 | 79.75 | 60.97 | 120 | 56 | 55 |
|       |     | 8:1 | 80 | 57.39 | 75.22 | 47.32 | 185 | 45 | 45.23 |
produced thinner, less rough polyvinyl alcohol layers with significantly lower permeability.

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