Elemental Characterization of the Topmost Atomic Layer of Surface Using Doppler Broadening Spectroscopy

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This work is dedicated to Prof. Peter S. Pershan on his 85th birthday. His many pioneering contributions to physics research have provided an inspiring model.

We present initial results that demonstrate the ability of Doppler broadening spectroscopy (DBS) to extract elemental information from the topmost atomic layer of a sample surface. Doppler broadening spectra from graphene grown on a polycrystalline Cu substrate was compared to the Doppler broadened spectra obtained after the partial removal of the bilayer graphene from the Cu surface. The Doppler broadening of the annihilation photopeak showed a clear difference between the as-inserted and the partially sputtered sample. We posit that this difference was due to the contribution of the gamma photons originating from the annihilation of positrons with the substrate (Cu) electrons to the total Doppler broadened spectrum. The Doppler broadened spectrum of the partially sputtered sample was well fit with a line shape produced by combining the Doppler broadened gamma spectrum of the as-inserted sample with the theoretically calculated annihilation gamma photo peak from bulk Cu. The fraction of the calculated Doppler broadened spectrum from bulk Cu in the fit spectrum is consistent with the increase in intensity of the Cu $M_{2,3}$VV Auger peak observed in the positron annihilation induced Auger electron spectra (PAES) of the partially sputtered sample. This suggests that analysis of the high momentum region of the gamma photopeak due to photons emitted following the annihilation of surface trapped positrons can provide a quantitative estimate of the elemental composition of the top surface. Our results provide evidence for the feasibility of using DBS as a new spectroscopic technique for the elemental characterization of the topmost atomic layer of surfaces.

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1. Introduction

The momentum of the annihilating electron–positron pair produces a deviation in the collinearity of the two annihilation gamma photons and Doppler shifts the annihilation photon energy as measured by the gamma detector [1]. The measurement of the angular correlation of two annihilation photons and the Doppler broadening of the annihilation gamma photopeak can therefore yield information about the momentum of the electron–positron pair. The momentum of the electron–positron pair is dominated by the momentum of the electron and therefore the analysis of the Doppler broadening provides chemical information of the site at which the positron annihilates [2]. Specifically, the high energy (“wing”) region of the photopeak, which is strongly affected by contributions from core electrons, has been widely used to obtain chemical environment information of the defects or interfacial surfaces in the bulk or subsurface region of the sample [3, 4]. In this paper, we provide evidence of the feasibility of using Doppler broadening spectroscopy for chemical characterization of the topmost atomic layer of exposed surfaces.

Previous studies have shown that positron annihilation induced Auger spectroscopy (PAES) can be used to determine the elemental content of the topmost atomic layer of surfaces due to the trapping of positrons in an image potential-induced surface state before annihilation [5]. This property has recently been utilized to obtain the PAES data from a single layer of graphene on a Cu substrate [6]. The wavefunction of the surface-trapped positrons decays rapidly and has appreciable overlap only with the electron wavefunctions spilling into the vacuum from the top surface. Weiss et al. [7] proposed that the gamma photon emitted following the annihilation of surface trapped positrons should provide highly surface selective chemical information and had shown that the high energy (equivalently high momentum) region of the annihilation gamma from multiple layers of a graphene sample on a Cu substrate was similar in shape to the high energy region of the theoretically calculated annihilation gamma from the bulk of graphite. Here, we extend the study further by comparing the

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Fig. 1. Schematic of the sample side of the advanced positron beam equipped with the time of flight spectrometer. Monoenergetic low energy positrons are deflected around the microchannel plate (MCP) via $E \times B$ drift resulting from the combined action of an axial magnetic field ($\sim 0.005$ Tesla) and transverse electric field generated by two sets of parallel plates. The positrons that exit the $E \times B$ system fly through a $\sim 3.0$ m field-free tube before reaching the sample. The electrons ejected following the positron impact and/or annihilation traverse the field-free tube before getting bent up into the electron detector. The annihilation gamma is detected using a high purity germanium detector (HPGe). The time difference between the detected gamma signal and the electron detection signal from the MCP provides the measured flight time of the electron. The energy of the annihilation gamma ray was determined through an analysis of the amplitude of the HPGe pulse.

Doppler broadening spectra from bilayer graphene grown on a Cu substrate to that obtained from the same surface after partial exposure of the Cu substrate by sputtering in order to investigate the ability of Doppler broadening analysis to estimate the atomic composition of the top atomic layer of exposed external or internal surfaces.

2. Experimental

The advanced positron beam, including the data acquisition system used for the present experiment, has been described elsewhere [8, 9] and only a brief description will be given here. A monoenergetic beam of low energy positrons are produced by a $^{22}$Na source housed inside of a source holder with a parabolic face onto which a thin film of solidified Ne is grown [10]. The low energy positrons emitted from the Ne film are guided to the sample using an axial magnetic field generated by multiple solenoid coils. As shown in Fig. 1, positrons are drifted around the microchannel plate (MCP) electron detector by an $E \times B$ field generated using a transverse electric field and the axial magnetic field. The positrons pass through a field-free tube after exiting the $E \times B$ system and before reaching the sample. The energy of the incident positrons on the sample can be varied from 2.5 eV to 20 keV by biasing the sample (from $-0.5$ V to $-20$ kV) with respect to the field-free tube.

Electrons may be emitted from the sample surface following the transition of the positron to a surface (or bulk) state and/or after the Auger decay of annihilation-induced holes. The emitted electrons travel through the field-free tube before drifting up into the MCP by the action of the $E \times B$ system, producing a signal associated with the detection of the electron. The annihilation gamma photons are detected by the high purity germanium (HPGe) detector that produces a corresponding voltage pulse. The time difference between the signal from the HPGe detector and the MCP is used to produce a histogram of the measured flight times of the electrons, whereas the amplitude of the HPGe detector pulse is used to produce a histogram of the gamma energy. The time-of-flight spectra shown in the manuscript has been constructed using digital data acquisition and analysis techniques [8], whereas the gamma spectrum has been constructed using conventional analog electronics. The HPGe pulse used for the construction of the gamma energy histogram by analog electronics also served as the trigger for the digital data acquisition, making the two spectra representative of the same surface features. The HPGe detector, with 11% efficiency, has an energy resolution of $\sim 1.1$ keV at the 511 keV photopeak and the timing spectrometer has a resolution of $\sim 22$ ns.

PAES and Doppler broadening spectra were collected from a bilayer graphene film grown on a polycrystalline Cu substrate purchased from ACS materials [11]. The sample was inserted into the vacuum chamber as received after which it was heated to between 375 K to 425 K in high-vacuum and then allowed to cool to room temperature before the data was collected. The measurements were repeated on the same sample after the partial removal of the graphene overlial by sputtering with Argon ions. A visual inspection of the sample revealed that the sputtering did not completely expose the underlying Cu substrate over the full region irradiated by the positron beam due to a partial
misalignment of the ion sputter gun. For the purpose of identification of the data in this paper, we will refer to the graphene on Cu sample before sputtering as “as-inserted” and to the same surface after sputtering as “partially sputtered.” The positron measurements were carried out at a sample chamber pressure of less than $6 \times 10^{-8}$ Pa.

3. Results and Discussion

Figure 2a and b show the PAES data collected from the sample surface before and after sputtering, respectively. The Auger spectrum from the sample before sputtering (Fig. 2a) was measured with a sample bias of $-0.5$ V with respect to the field-free tube, and the maximum kinetic energy of the positrons that were incident of the sample was $\sim 7$ eV. The PAES data from the partially sputtered sample was collected with $\sim 42$ eV positrons. We increased the kinetic energy of the positrons incident on the partially sputtered surface to improve the count rate and thus, to reduce the total data collection time maintaining the surface chemical composition of the partially sputtered surface during the experiment. In this setting, the field-free tube was also held at $-43$ V relative to the sample which prevents electrons with energy less than 43 eV from reaching the electron detector.

The Auger spectra from the sample before sputtering has three main peaks: the peak corresponding to the M$_{2,3}$ VV Auger transition from Cu, the KVV Auger peak from C, and the KVV Auger peak from O. The presence of Cu in the as-inserted sample is attributed to small, but finite, overlap of the positron wavefunction with Cu atoms underneath the graphene overlayer [6]. After the partial sputtering of the bilayer graphene overlayer, all three Auger peaks remain. However, the intensity of the Cu M$_{2,3}$ VV Auger peak measured relative to the intensity of the KVV Auger peaks from C and O increased by $\sim 15\%$.

The Doppler broadening spectra collected from the as-inserted and partially sputtered sample is shown in Fig. 3a along with the Doppler broadening spectrum representing the bulk of the Cu substrate. A background has been subtracted from all the gamma spectra following the prescription given by Mogensen et al. [12]. The “bulk”-like spectrum was collected by biasing the sample to $-20$ kV with respect to the field-free tube, whereas the Doppler broadening spectrum from the as-inserted and partially sputtered sample surfaces was measured with the settings described above for PAES measurements. Particularly, the Doppler broadening spectra representing as-inserted and partially sputtered surfaces were collected simultaneously with the PAES spectrum, though not in true coincidence [8]. The voltage pulse from the HPGe detector used for the construction of gamma energy histogram using the analog electronics also served as the start signal for the time-of-flight measurement using digital data acquisition techniques. The Doppler broadened annihilation gamma spectra from the surface of the sample (bilayer graphene and partially sputtered surface) contains contributions from positrons annihilating in the surface state and from annihilations in the singlet or triplet Ps state in vacuum with little contribution from bulk of the sample. On the other hand, the Doppler broadening data collected with $-20$ kV on the sample predominantly contain bulk state annihilations with less than 3% of the total data collected resulting from surface state or bound
The Doppler broadening spectrum from bilayer graphene is well described by the calculated spectrum from graphite, especially in the high momentum region. In contrast, the high momentum region or the “wing” of the Doppler broadened annihilation spectra from the partially sputtered surface and the spectra collected using 20 keV positrons progressively broadens towards the wing region of the calculated spectrum from bulk Cu. In order to amplify the features in each Doppler broadened spectrum a ratio curve analysis has been performed; the results of which are shown in Fig. 3b. The ratio of the experimental Doppler broadening spectra from different surfaces was taken with respect to the Doppler broadening annihilation spectrum measured using 20 keV positrons. The calculated spectrum from bulk graphite has been divided by the calculated Cu spectrum to obtain the theoretical ratio curve. The ratio curve of both the as-inserted bilayer graphene and the partially sputtered surface is different from the theoretically calculated ratio at low momentum. However, at higher momentum ($> 10^{-3} m_0 c$), the shape of the as-inserted surface is very similar to that of bulk graphite, showing the presence of graphitic carbon on the “as-received” surface. The difference at the low momentum can be due to (i) the difference in bulk and surface projected valence electronic density and (ii) the fact that Doppler broadening spectrum from 20 keV does not entirely represent the defect-free bulk Cu spectrum due to the presence of defects in the polycrystalline Cu substrate and small contribution from surface state/o-Ps pick off annihilations. The higher momentum region of the ratio curve from the partially sputtered surface shows a clear trend towards unity, signalling the presence of annihilations with $3d$ electrons in Cu; the absence of which in the graphitic sample is responsible for the dip seen at $\sim 15 \times 10^{-3} m_0 c$ [14]. These results show that the high momentum region of the Doppler broadening spectrum reflects the surface chemical composition even when the spectra contain other features that affect the line shape in the low momentum region.

In order to determine the relative increase in annihilations with Cu atoms, the Doppler broadened annihilation spectrum from partially sputtered surface was fit using an empirical annihilation gamma line shape generated by combining the experimental Doppler spectrum from as-inserted bilayer graphene sample with the Doppler spectrum calculated from bulk Cu. The ratio of the empirical curve (85% the spectrum from the as-received sample and 15% the calculated Cu spectrum) taken with respect to the Doppler broadening annihilation spectrum measured using 20 keV positrons is shown in Fig. 4 along with the ratio curve from the as-inserted surface and the partially sputtered surface. The fit captures most of the line shape of the partially sputtered surface especially the dip seen at $\sim 15 \times 10^{-3} m_0 c$. The intensity of the calculated Cu spectrum in the fit curve is consistent with the increase in intensity of the Cu $M_{2,3}$V Auger peak observed in the PAES spectrum from the partially sputtered surface. The experimental annihilation gamma photo peak has been compared to the theoretically calculated Doppler spectrum from bulk of graphite and Cu in Fig. 3a. Details of the calculation have been discussed in detail in previous publications [13]. The theoretical gamma spectra have been convoluted with a Gaussian representing the energy resolution of the detector at 511 keV.
sputtered surface. These results demonstrate the ability of Doppler broadening spectroscopy to provide relative concentration of the elemental species at the topmost atomic layer of the surface.

Presently there is a need for a spectroscopic technique that can selectively address the surface modification on the inner surface of porous materials. The existing surface science techniques based on electrons, photons, or neutrons cannot provide direct surface selective information from the top layer of the inner surfaces of nano and micro porous materials. One of the reasons is that the escape depth of the excited electrons from which conventional surface analytic techniques derive their spectroscopic information is too shallow to investigate internal porous surfaces. Most other probes (photons or neutrons) provide information that is averaged over a range of depths and hence, is not surface selective enough when the pore ligaments are only a few nanometer thick. In contrast, positrons have a high affinity for trapping at these internal porous surfaces and the annihilation gamma rays can escape through thick samples and sample chambers carrying chemical information from the inner surface. We have shown above that the high momentum region of the annihilation gamma photopeak can provide a quantitative estimate of the elemental composition of the topmost atomic layer of external surface.

We posit that similar chemically sensitive information will be contained in the annihilation gamma photopeak even when they are escaping from the inner surfaces of porous materials. As a result, Doppler broadening spectroscopy can become a unique technique of characterizing the internal chemistry of the topmost atomic layer of these internal surfaces.

4. Conclusion

Doppler broadened annihilation gamma spectrum from bilayer graphene grown on a polycrystalline Cu substrate was compared to the Doppler broadened spectra obtained after the partial removal of graphene overlayer from the Cu surface. The Doppler broadened gamma spectrum from the partially sputtered surface shows a marked difference from the spectrum obtained from the bilayer graphene in the high momentum region due to annihilation with electrons belonging to the surface Cu atoms. Using an empirical fit to the gamma spectrum from the partially sputtered surface an estimate of the increase in Cu annihilations relative to the as-inserted sample was obtained. The obtained estimate was found to be consistent with the increase in intensity of the Cu M$_{2,3}$VV Auger peak observed in the PAES data of the partially sputtered sample. Thus, an analysis of the gamma photopeak emitted following the annihilation of surface trapped positrons can provide a quantitative estimate of the elemental composition of the top surface. Based on these results, we propose that Doppler broadening spectroscopy can be used as an effective in-operando spectroscopic tool for the elemental characterization of the topmost atomic layer of the external surfaces of thin films and inner surfaces of microporous and nanoporous materials. Doppler broadening spectroscopy can fill in the existing lacuna in surface science for the characterization of the inner surfaces of porous materials.

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