High-quality microresonators in the long-wave infrared based on native germanium

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The longwave infrared (LWIR) region of the spectrum spans 8 to 14 μm and enables high-performance sensing and imaging for detection, ranging, and monitoring. Chip-scale LWIR photonics has enormous potential for real-time environmental monitoring, explosive detection, and biomedicine. However, realizing technologies such as precision sensors and broadband frequency combs requires ultra low-loss and low-dispersion components, which have so far remained elusive in this regime. Here, we use native germanium to demonstrate the first high-quality microresonators in the LWIR. These microresonators are coupled to partially-suspended Ge waveguides on a separate glass chip, allowing for the first unambiguous measurements of isolated linewidths. At 8 μm, we measured losses of 0.5 dB/cm and intrinsic quality (Q) factors of 2.5 × 10^5, nearly two orders of magnitude higher than prior LWIR resonators. Our work portends the development of novel sensing and nonlinear photonics in the LWIR regime.

Optical microresonators are an essential tool for nonlinear optics. By storing high energy in a small volume, they significantly enhance nonlinear light-matter interactions. A number of interesting effects can be demonstrated in them: frequency comb generation 1-2, ultrasensitive sensing 3-4, optomechanics 5-6, and non-Hermitian physics 7-9, to name a few. In the near-infrared, platforms such as SiO2 microtoroids 10 and Si3N4 ring resonators 11 can have Q factors of billions, while at longer-wavelength midwave infrared, Q values can be sustained at the level of several million 12. Still, there is significant interest in pushing to even longer wavelengths. The rovibrational transition of molecules in the LWIR can be orders of magnitude stronger than shorter wavelengths 13-16, and of course most blackbody radiation occurs in this range, which is why forward-looking infrared imaging has become so prominent.

While high-power sources like CO2 lasers and quantum cascade lasers (QCLs) 17-21 are readily available at LWIR, chip-scale nonlinear photonics remains challenging. The scaling with the wavelength becomes unfavorable and almost all materials become lossy. Chalcogenides become moderately lossy beyond 7 μm 22, and silicon has significant absorption loss over 1 dB/cm beyond 7 μm due to multi-phonon absorption 23-25. So far, the lack of high-Q microresonators and correspondingly low-loss waveguides suitable for LWIR has remained a bottleneck for LWIR photonics. Despite excellent etch profiles and high-quality epitaxy 26-28, none have even scanned a laser across a resonance and observed clear features much narrower than the free spectral range, something routine at shorter wavelengths. Germanium is a legendary material platform in which the first transistors were realized, and it is fairly compatible with mainstream silicon technology 29. With the highest refractive index (n₀ = 4), a large nonlinear refractive index (n₂ = 20 nm²/W) 30, and a wide transparency window (from 2 to 14 μm), Ge shows excellent potential as a low-loss material platform at LWIR 30,31. However, fabrication of Ge-based on-chip photonic devices also requires compatible cladding materials that confine the light. Recently, epitaxially-grown Ge-on-GaAs and Ge-on-Si has been used to make integrated waveguides and ring resonators 32-37. The optical mode was primarily confined in the Ge part, achieving waveguide losses of 4 dB/cm at LWIR. Specifically, a 4 μm wide Ge-on-Si rib waveguide showed losses of 2.7 dB/cm at 8 μm and 1 dB/cm at 10 μm, while a graded SiGe waveguide showed losses of 3 dB/cm at 8 μm 38,39. However, these losses are significantly higher than the intrinsic losses of high-quality Ge grown by the traditional Czochralski crystal pulling method under thermodynamic

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equilibrium conditions. In fact, the losses of pure germanium are so low (<0.15 cm⁻¹) that high-quality measurements of the losses were only performed recently⁴⁰. Although epitaxial Ge can be atomically smooth, whenever GaAs or Si are used as the substrate for heteroepitaxy, inevitable degradation of the LWIR transparency results. For Ge grown on GaAs, the volatile arsenic background and the Ga interdiffusion causes unwanted impurity incorporation, lowering the optical transparency at the LWIR⁴¹. Although Ge grown on Si will not be impacted by the unwanted background impurity incorporation as in GaAs, there is a significant lattice mismatch between Si and Ge. This results in a high density of threading dislocations and significantly lower material quality⁴². The thick waveguides needed for the LWIR exacerbate this issue, and the high index of silicon allows light to penetrate into silicon.

In this work, we utilize the highest-quality, lowest-loss Ge material in conjunction with mechanical fabrication processes to fabricate extremely high-Q whispering gallery mode (WGM) resonators and low-loss waveguides. The structures are initially bonded to anodic glass substrates (essentially creating thick Ge-on-insulator), but the glass adjacent to Ge is mostly removed, resulting in structures that are ultimately overhung. These microresonators have ultra-smooth surfaces and are easily separated from their substrates using conventional etchants. To evaluate the optical properties of the resonator, we perform transmission measurements by focusing the emission of an 8 μm distributed feedback (DFB) QCL into a semi-suspended Ge waveguide, which is coupled into the microresonator. By sweeping the wavelength of the QCL, a loaded Q₂ of 2.2 × 10⁸ and an intrinsic Q₂ of 2.5 × 10⁸ are measured. This is similar for both the transverse electric (TE) and transverse magnetic (TM) polarizations.

**Results**

**Fabrication process and devices**

Figure 1a–d shows scanning electron microscope (SEM) images of the fabricated devices. A microresonator with diameter of 450 μm can be seen in Fig. 1a. The resonator has a uniform thickness across the whole resonator, while the outermost 100 μm is suspended away from the dune-shaped glass supporting structure in the center. This eliminates the losses due to the glass substrate at LWIR. To look at the active part of the WGM microresonator, we zoomed in to the edge of the WGM microresonator. Figure 1b shows a zoomed image of the edge. It is angled inward by about 30-degrees, a consequence of isotropic etching and edge rounding from the polishing process. By changing the plasma strength with controlled anisotropic etching and tuning the polishing pressure, the geometry could be further tuned, with potential for dispersion engineering⁴³. To efficiently couple LWIR light into the WGM microresonator, we fabricated a partially-suspended Ge waveguide on a glass chip with dimensions of 4 × 2 mm², shown in Fig. 1c. The waveguide is around 5 μm thick, similar to that of the microresonator, and is 10 μm wide. The curved part of the suspended waveguide shows uniform geometry and smooth surface, making it ideal for coupling measurements.

Figure 1e shows an overview of the fabrication process of the microresonators and of the partially-suspended waveguides. The details are summarized in the Methods, but briefly it relies on a combination of etching and post-polishing. The bottom side of the WGM microresonator has only gone through the wafer bonding process and has the same surface roughness as the original polished Ge wafer, and we did not find a significant difference in the surface roughness between the side-facet and the bottom facet. We performed atomic force microscopy in tapping mode in the center of the WGM microresonator to evaluate the surface roughness of polished Ge surface. Only sparsely distributed spot-like extrusions with the height of tens of nanometers could be found over the 13.5 × 13.5 μm² area (Supplementary Fig. S2). The surface roughness of polished WGM microresonator is about 2 orders of magnitude smaller than the wavelength in LWIR, so we can safely ensure that there is no significant surface roughness-induced scattering loss for our WGM microresonator at the LWIR. In addition, we show in Fig. 1d a fabricated Ge microresonator waveguide (Supplementary Fig. S3). The waveguide is on the left and the glass substrate is on the right. d SEM images of a WGM microresonator array. e Overview of the fabrication procedure, which relies on a combination of wafer bonding and polishing processes.
array, proving the versatility and robustness of the non-epitaxial Ge platform.

**Optical simulation**

To simulate the light propagation in the microresonator, the waveguides and their coupling behavior, we performed finite element method (FEM) simulations based on the geometry and dimensions shown in Fig. 1. The effective refractive indices of 3.88 and 3.89 were calculated for the fundamental quasi-TE mode in the WGM microresonator and Ge waveguide, respectively, with each mode profile shown in Fig. 2a. To fully explore the potential of LWIR Ge photonics at 8 μm, we calculated the free spectral range (FSR, a.k.a. $D_0$) and the change in FSR per mode ($D_0$), which are commonly used to characterize microcavities for nonlinear optics. They are defined by

$$\omega_n = \omega_0 + D_0 \mu + \frac{1}{2} D_2 \mu^2 + \ldots$$  (1)

where $\mu$ is the mode number and $\omega_n$ is the $n$th frequency. The calculations were performed using the simulations of WGM microresonators with thicknesses ranging from 2.5 μm, the typical thickness of epitaxial Ge-on-Si, up to 12.5 μm. As is shown in Fig. 2b, there is a significant increase of the FSR with increasing thickness of the WGM microresonator (over the entire spectrum LWIR range). Similarly, with increasing thickness $D_2$ goes from positive to negative. This tuning range is several orders of magnitude higher than that of the SiO$_2$ WGM microresonators, a result of the significant refractive index contrast between Ge and air. Note that achieving low and flat anomalous dispersion with these resonators requires Ge thicknesses in the 7.5-10 μm range, which is very difficult using epitaxially-grown samples on non-native substrates. To further explore the coupling, we used coupled mode theory together with the simulated mode profiles in Fig. 2a to calculate the coupling between waveguide and disk (top of Fig. 2c) as a function of waveguide position. As the waveguide is moved closer to the resonator mode, the coupling increases drastically. We also calculate the transmission through the waveguide when it is coupled to a cavity with intrinsic modal losses of .13 cm$^{-1}$ (bottom of Fig. 2c). When the waveguide is close to the disk mode, the external coupling losses match the internal losses and critical coupling is achieved, causing the transmission to become zero.

**Optical setup and measurements**

To characterize the microresonators, we performed direct transmission measurements using a single-mode DFB QCL. The wavelength of the QCL can be tuned from 1285.5 to 1290.5 cm$^{-1}$ by controlling both the QCL current and temperature. As is shown in the schematic of the optical path and components in Fig. 3a, the collimated laser beam was expanded through two off-axis parabola (OAP) pairs by a factor of ten and two times, consecutively, reaching a beam size of about 30 mm. Next, the collimated beam was focused on the fabricated Ge waveguide entrance facet using 1 inch OAP mirror, photographed in Fig. 3b, reaching a beam size of about 20 × 20 μm$^2$. The transmitted laser signal from the waveguide was collected by an OAP of 4-inch focal length, and the collimated beam size was reduced by two times using a pair of OAPs. The transmitted signal was collected using a Mercury-Cadmium-Telluride (MCT) detector connected to a lock-in amplifier system to eliminate background. We further used two pinholes at the confocal planes of the OAP mirror pairs to exclude unwanted interference from the scattered laser signal in the system. These were aligned visually using an electronic autocollimator. In conjunction with the fact that the waveguide is bent by 90 degrees (see Supplementary Fig. S3 for a more detailed picture), no scattered light was observed. The waveguide chip was mounted on a piezoelectric actuator that allowed fine adjustment of the distance between the waveguide and the resonator.

Figure 3d shows a pair of representative scans of waveguide with and without the coupling to the WGM microresonator. In the reference scan, the three dips of the transmission signals match well to the H$_2$O gas lines at 1287.40 cm$^{-1}$, 1288.25 cm$^{-1}$, and 1288.89 cm$^{-1}$ (Supplementary Fig. S4). We use these three gas absorption features as fine calibration for the QCL wavelength. When the WGM microresonator is coupled to the waveguide, we observed a series of sharp dips much narrower than the three H$_2$O gas absorption lines. These come from coupling between the waveguide and the WGM microresonator. To acquire the transmission information without interference from the signal variation of the background, we divide the signal by the reference; the normalized transmission is plotted in Fig. 3e. The most prominent dip corresponds to coupling between the waveguide and the WGM microresonator. To acquire the transmission information without interference from the signal variation of the background, we divide the signal by the reference; the normalized transmission is plotted in Fig. 3e. The most prominent dip corresponds to coupling between the waveguide and the WGM microresonator.
To investigate the coupling condition, we performed a series of transmission measurements by changing the voltage of the piezoelectric actuator, which in turn changes the separation between the WGM microresonator and the waveguide. Figure 4d shows the transmission versus piezoelectric actuator voltage for TE polarization, and we can see a monotonic decrease of the transmission by reducing the voltage of the piezoelectric actuator. If we convert the voltage change to the distance of movement by estimating travel per distance ratio of $0.2 \, \mu m/V$ of the piezoelectric actuator, we can compare this to the simulations in Fig. 2. From this plot, we can infer that the 60% transmission dips are in an under-coupled condition. Those additional transmission dips with smaller coupling strengths correspond to coupling with higher-order modes, which start to grow significantly as the waveguide is brought closer (Supplementary Fig. S5), suggesting parasitic coupling between the waveguide and higher-order microresonator modes\textsuperscript{46}. These mode interactions can induce anomalous GVD in normal dispersion devices and lower the threshold for Kerr frequency comb generation\textsuperscript{47–50}. Further development of a narrower single-mode coupler will enable coupling at a greater
TE Polarization losses to be 9 cm. The suspended portion of the waveguides is likely similar, although the values are possible for this material (Supplementary Fig. S7). The loss of cm). This value is very close to the reported losses of germanium in this previous microresonators in the LWIR, which are summarized in Table 1. These values are almost two orders of magnitude higher than those from microresonators-waveguide spacing. The intrinsic Q factor can be calculated using27:

\[ Q_i = \frac{2Q_f}{1 + \sqrt{T}} \]  

where \( Q_i \) is the loaded Q factor and \( T \) is the transmission when undercoupled. Using this, we calculate intrinsic Q factors of 251,000 and 253,000 for the TE and TM polarization, respectively. There is little difference between these \( Q_s \), suggesting that all surfaces of the WGM microresonators are equally smooth. We fitted 12 of the transmission dips at 1285.38 and at 1287.06 cm\(^{-1}\) in undercoupled condition, and the averaged Q factor is \( 2.2 \times 10^4 \) (Supplementary Fig. S6). These high-Q values are almost two orders of magnitude higher than those from previous microresonators in the LWIR, which are summarized in Table 1.

**Discussion**

From these intrinsic \( Q_s \), we can now directly estimate the effective loss of the mode using \( a = \frac{2\pi n_e}{\lambda_0} \), giving an intrinsic loss of 0.13 cm\(^{-1}\) (0.5 dB/cm). This value is very close to the reported losses of germanium in this range, showing that these resonators have the highest quality factors that are possible for this material (Supplementary Fig. S7). The loss of the suspended portion of the waveguides is likely similar, although the larger losses of the glass-clad portion makes the averaged measured losses to be 9 cm\(^{-1}\) (Supplementary Fig. S3). At even longer wavelengths, beyond the onset of Ge’s three-phonon absorption, it may be beneficial to further explore alternative materials such as diamond and KBr using suspended structures. Still, Ge is quite competitive in these wavelengths due to its combined low loss and fabrication feasibility. Integration of Ge-on-diamond could also be an appealing combination for monolithic low-loss Ge waveguides.

With this development, many of the microresonator-based technologies that were previously not viable in the LWIR are now feasible. For example, an attractive application for this platform is for chip-scale ultrabroadband sources pumped by QCLs, like Kerr comb generation (in the resonator geometry). While combs can be directly generated by QCLs\(^{15,16,21,22}\), achieving spectral bandwidths over 100 cm\(^{-1}\) is very challenging. The extremely wide transparency range of germanium makes it attractive for this application, as with it octave-spanning operation could be feasible. While in this work we did not attempt Kerr frequency comb generation as the inefficient coupling into and out of the waveguide precluded this, it is worth calculating the parametric threshold for frequency comb generation. Assuming a modal diameter of 5 μm, we calculate that the threshold for comb generation should be approximately 240 mW under optimal conditions\(^{21}\). This threshold is well below the 10 W range produced by state-of-the-art LWIR QCLs\(^{21}\). Still, achieving fully-integrated comb generation in the current implementation of this platform would be difficult due to the suspension required even for the waveguide. Though the waveguides are fairly lossy when on the glass, they are actually remarkably low when one considers that the losses of glass can exceed 10,000 cm\(^{-1}\) in the
LWIR. This is a result of the large index mismatch of germanium and the glass, leading to a small overlap of the mode with the lossy region. Predeposition of a thin film of a moderately low-loss low-index mater-
rial, such as ZnS, would drastically reduce the loss to a similar level as the disks (Supplementary Fig. S3).

In conclusion, we presented a non-epitaxial fabrication platform for germanium microresonators and waveguides whose quality factors are two orders of magnitude higher than prior work. This methodology avoids the defects inevitably associated with thick epitaxial growth. Using a single-mode DFB QCL, we observed transmission dips with repetition rate matched well to the FSRs of the WGM microresonators. Loaded Q factors of 221,000 and 224,000 were measured for the TE and TM polarization, respectively, corresponding to the intrinsic Q factors of 2.5 × 10^5. These excellent Q values represent a milestone for the low-loss LWIR photonics, and we expect similar platforms to be essential for future chip-scale sensing and nonlinear photonics.

Methods
Wafer bonding process
A sample of Hoya SD-2 anodic bonding glass of 1.5 × 1.5 cm^2 in size and 1 mm in thickness was firstly cleaned by acetone and isopropanol. This was followed by a hot bath in H_2O_2:CH_3OH:H_2O (volume ratio of 1:1:5) at 90 °C for 15 min. This step ensures that the surface of the glass has high surface energy, favorable for hydrophobic bonding. A piece of highly resistive Ge of 1 × 1 cm^2 in size and 500 μm in thickness was dipped into the same H_2O_2:NH_4OH:H_2O solution (at room temperature for 5 s). Any surface treatment of Ge at a higher temperature or for a longer time will lead to surface degradation. After the surface treat-
ment, the glass and Ge piece were sandwiched into graphite plates inside a thermocompression bonder. 3000 N force was applied, and the bonder was heated to 300 °C for 60 min (Supplementary Fig. S1).

Nanofabrication process
After the mechanical lapping and polishing process, which utilized adhesive lapping to minimize scratch formation, electron beam litho-
graphy (EBL) was used to define the features of the Ge, using a Vistec EBPG5200 EBL System at 100 KV. HMDS adhesion promoter and Spacer-300Z discharger were applied before and after the spin-coating of the e-beam resist MA-N-2403. This step is essential. An SF_6 chemistry was followed by a hot bath in H_2O_2:NH_4OH:H_2O (volume ratio of 1:1:5) and then the sample was etched back by using the same SF_6 chemistry. The curved Ge waveguides were fabricated using the same process in order to efficiently couple the light from free space into the microresonator. The Ge waveguide has a larger in-plane end facet to enhance the collection of the light from the free space. Both the coupling sections of the Ge microresonator and the waveguide were diced to the edge of the glass chips (Supplementary Fig. S8 and Section VI). A final HF etching process was used to further etch back the glass, suspending both the Ge microresonator and the Ge waveguide from the glass chips.

Tuning of the quantum cascade laser
A DFB QCL with peak power of 250 mW and wavelength tuning range from 1283.5 to 1290.5 cm\(^{-1}\) was used to perform the wavelength-dependent transmission measurements. A linear parameter space of current and temperature was established in order to wavelength-tune at a constant power of 10 mW. An Arroyo 6310-QCL controller was used to control the temperature of the TEC and current of the QCL laser, with a minimum step size of 0.01 °C and 0.02 mA.

Data availability
The data and code that support the findings of this study are available in Figshare with the identifier https://doi.org/10.6084/m9.figshare.20369253.

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Author contributions
D.R. and D.B. initiated the project. D.R. fabricated the devices and performed the measurements. D.R. and D.B. designed the measurement setups and performed simulations. D.R. and C.D. performed the experiments. S.A. grew III-V samples used for developing hydrophilic
bonding protocols. D.R. wrote the manuscript with contribution from all authors. D.B. supervised the whole project.

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**Competing interests**
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