Poly(melamine-formaldehyde-silica) Composite Hydrogel for Methylene Blue Removal

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Received: December 16, 2020; Revised: March 28, 2021; Accepted: May 11, 2021

In this work, we presented the synthesis and characterization of poly-melamine-formaldehyde-silica (PMF-Si) composite from poly-melamine-formaldehyde (PMF) and previously purified commercial sodium silicate used as a precursor of silica nanoparticles (Si-NPs). The materials were characterized by several physicochemical techniques such as XRD, FTIR, SEM, TEM, AFM, TGA, and DTG analysis to access their properties. We also studied hydrogel dehydration, rehydration, the potential of zero charge (PZC), and the adsorption capacity of the dye methylene blue (MB) in the hydrogel. From XRD analysis, PMF and PMF-Si materials presented amorphous profiles. AFM, SEM, and TEM images revealed a fibrous PMF structure and homogeneous dispersion of Si-NPs through the PMF-Si composite. The Si-NPs exhibited an average diameter between 10 to 15 nm. The first PMF-Si hydration reached 93% after synthesis, and complete rehydration is possible up to 70% of dehydration. PMF-Si has the PZC at pH = 1.22 and, at superior values of pH, the composite exhibits superior adsorption rates of MB. The adsorption tests performed to remove MB dye using PMF-Si hydrogel showed a q_{max} of 140 mg/g, which is superior to other materials routinely used for MB dye removal.

Keywords: Melamine-formaldehyde-silica, Hydrogel, Methylene blue, Dye removal.

1. Introduction

Hydrogels are composed of three-dimensional (3-D) insoluble polymer materials containing hydrophilic groups (modified or synthetic), capable of absorbing and retaining a considerable amount of water due to their morphological expansion1-3. The increasing hydrogels synthesis methods aim to supply the demand for biocompatible materials and adequate for use in different applications such as pollutant removal, development of medicines, cosmetics, hygiene, and agricultural products23. Hydrogels are demanded because they present unique properties that account for biodegradability, biocompatibility, low-cost, facile synthesis, renewability, and excellent hydrophilicity4-7.

The use of polymers in the synthesis of hydrogels aims to increase hydrophilic capacity4. Poly(N-isopropyl acrylamide)8, poly(ethylene glycol)9, poly(vinyl alcohol)10, poly(cellulose/chitosan)11, and polyurethane12 are examples of polymers extensively used in hydrogel fabrication. These polymers have in common hydroxyl and nitrogen groups capable of establishing hydrogen or van der Waals bonds through their structures, enabling them to absorb significant amounts of water4.

Poly-melamine-formaldehyde (PMF) polymer exhibits advantages that allow its use in hydrogels synthesis because of its large amounts of nitrogen, high resistance to chemical attack, and low-cost of production13. Besides, PMF offers a comprehensive 3-D structure that can be easily broken down; it can be applied in liquid filtration, sound, thermal and electrical insulation, development of components for construction industries, packages, high-quality laminated surfaces, fire retardant, heavy metal removal, dyes adsorption, etc14-17. The methylol monomers exhibit attractive properties like low molar weight and up to 9 different possibilities of intermolecular interactions comprising 6 hydroxyl groups and 3 nitrogen atoms18. The numerous possible points of interaction make PMF suitable for hydrogel fabrication13.

Applications of PMF-based hydrogels are mentioned in recent studies19-22. Some of those works highlight new composites’ development using synthetic or modified silica nanoparticles (Si-NPs) incorporated into PMF23-26. Si-NPs are indicated because of their high-active surface and high-density silanol groups27. Schwarz and Weber synthetized a xerogel by incorporation of Si-NPs into PMF aqueous solution28. They also synthesized a mesoporous poly(melamineformaldehyde-silica) (PMF-Si) composite through reverse emulsion polymerization29.
The decision to incorporate Si-NPs into the PMF is due to beneficial properties, e.g., high surface area and available silanol groups, allowing PMF to cluster through intermolecular interactions, culminating in chemical and thermal stability, with applications like catalyst, drugs, and adsorption of dyes. Liu et al. reported the synthesis of a new material in which they used melamine-formaldehyde as a skeleton to stabilize the silica-monolith sorbent. Then, the authors found that the synthesized material has a high surface area and tunable surface properties. Liu et al. prepared mesoporous silver-melamine nanocomposites that presented a high adsorption capacity of Hg(II) as 589.99 mg/g. Research like Liu and Li shows how promising composites’ development is based on the melamine-formaldehyde polymer to obtain materials with better absorption properties, combined with low-cost and ease preparation.

Specifically, the generation of dye waste draws attention because it causes serious harm to humans; they behave as toxic and carcinogenic residues and present high stability to light, temperature, detergents, and microbial attack. Several types of dyes are applied in textile, leather, paint, food, cosmetics, and paper industries. Adsorption is a suitable method for metal ions and dye removal from the wastewater due to its low cost and simplicity compared to other procedures. Heavy metals and dyes have been removed appropriately with the use of adsorbents obtained from chemical modification. MB is a crucial azo dye commonly used in the industry. However, MB dye causes severe damage to human health, including difficulty breathing, nausea, cyanosis, jaundice, quadriplegia, and tissue necrosis.

Additionally, MB can impair photosynthetic processes in aquatic ecosystems. Various chemical, physical, and biological methods have been applied to remove dyes from wastewater, including chemical coagulation/precipitation, flocculation, ozonation, oxidation processes, ion exchange, and ultrafiltration. However, they can have considerable disadvantages such as high cost and generation of secondary pollutants. Adsorption of dyes such as MB is a method that exhibits advantages over other methods in terms of efficiency, design, cost, and availability, especially when the dye concentration is not high.

MB is commonly used for dyeing various types of materials, such as cotton, wool, and paper. MB’s use causes dye residues in waters that need to be treated before being thrown into the environment. Therefore, there is an urgent need to develop non-toxic and inexpensive new materials to eliminate dye pollutants from water. Like PMF-Si are eco-safety choice for wastewater dye removal. Therefore, among other applications for PMF-Si, the MB adsorption is of relevance in the present study.

Thus, using consolidated synthesis protocols through the sol-gel method of synthesis, the present work aims to synthesize a PMF-Si hydrogel via a low-cost procedure, using commercial sodium silicate solution as the precursor of Si-NPs. Furthermore, we used the synthesized hydrogel to remove methylene blue dye (MB). The main differential of this research is the formation of a composite (PMF-Si) in which Si-NPs are formed within the structural network of the PMF, simplifying the composite preparation process and eliminating the cost of acquiring commercial Si-NPs.

2. Experimental

2.1. Sodium silicate purification

Unpurified sodium silicate solution (also containing 22.66 wt.% of water, 9.2 wt.% of sodium hydroxide, and 0.8 wt.% of iron) was purchased in a local city market in Teresina-PI, Brazil. We purified the sodium silicate material before using it in later stages. The previous purification procedure is recommended for impurity removal, such as iron ions. We used hydrochloric acid to solubilize iron precipitates forming their ions. The silica purification process consisted of using 200 g (120 mL) of the commercial sodium silicate solution diluted to 1 L of deionized water at 25 °C. Then, 28 mL of concentrated hydrochloric acid (Vetec 32%) was added dropwise while the solution was being stirred. The chemical Equation 1 describes the addition of hydrochloric acid to the diluted sodium silicate solution:

\[
\text{Na}_2\text{SiO}_3 + 2 \text{ HCl} \rightarrow \text{SiO}_2 + 2 \text{ NaCl} + \text{ H}_2\text{O}
\]

During the hydrochloric acid addition, there was precipitation of amorphous silica, and the pH value decreased to 8.0. The precipitate was filtered and washed with deionized water. The purified silica material was oven-dried at 65 °C for 24 h.

Finally, we added 135 g of the purified silica to the 4.5 mol/L NaOH (Vetec 97%) solution under stirring at 70 °C, forming 1 L of sodium silicate solution in the concentration of 274 g/mL, pH = 14.0. The conversion of silica to sodium silicate follows the chemical Equation 2:

\[
\text{SiO}_2 + 2 \text{ NaOH} \rightarrow \text{Na}_2\text{SiO}_3 + \text{ H}_2\text{O}
\]

The purified sodium silicate solution was stoked for posterior use in the hydrogel formation to form Si-NPs.

2.2. Melamine-formaldehyde polymer synthesis

We have conducted the synthesis of PMF according to the work of Merline et al. The reaction between melamine and formaldehyde took place at a molar ratio of 1:3, and the polymerization mechanism of melamine-formaldehyde resin is known from the literature. The initial procedure consisted of the dissolution of 1.5 g of melamine (Sigma Aldrich 99%) in 25 mL of deionized water (18.2 MΩ cm) in an open system. The solution was stirred and heated at 70 °C, followed by the addition of 3 mL of formaldehyde (Merck 38%), and kept under agitation to complete the PMF formation. The shift of white color to a translucent solution indicated that the reaction had happened. Later, we maintained the system with the exact condition of stirring and heating, and 30 mL of glycerin (Vetec 99.5%) was added and stirred for another 10 min. The glycerin consists of a source of hydroxyl groups to the system, which is susceptible to make hydrogen bonds allowing the formation of a structure capable of absorbing large amounts of water. Therefore, the final solution consists of the mixture of the PMF and glycerin used in the hydrogel synthesis.

2.3. Hydrogel synthesis

The silicate in acid medium forms hydrogels with Si-NPs and, depending on the proportion between siloxane (Si-O-Si) and silane (Si-OH) groups, and it can have more or less hydrophilic character. The PMF provides the composite
sustainment because it is a fibrous structure, one of the most important findings in this work. 280 mL of the previously prepared sodium silicate solution (274 g/L) was added in a beaker with 670 mL of deionized water, as described in section 2.1. The diluted sodium silicate solution was heated to 70 °C and kept under stirring. Then, we added the previously prepared PMF-glycerin solution, as described in the previous section. After adding the PMF-glycerin solution, the mixture was stirred for 5 min, followed by the dropwise addition of 10 mL of concentrated acetic acid (Vetec 99.5%) to allow the formation of Si-NPs in the hydrogel. After the addition of acetic acid, we found the hydrogel pH = 7.5. Figure 1a represents the stages of formation of the PMF-glycerin resin, while Figure 1b shows the formation of the PMF-Si composite. The original hydration of the PMF-Si composite was 93%, determined after the preparation procedure.

Figure 1. (a) Simplified depiction of PMF synthesis and incorporation of glycerin. (b) Preparation of the PMF-Si composite.
2.4. Material characterizations

The synthesized hydrogel PMF-Si was submitted contact per time 48 h with sulfuric acid (4.5 mol/L), nitric acid (4 mol/L), phosphoric acid (4 mol/L), methyl alcohol, ethyl alcohol, acetylene, and benzene solutions at 25 °C. However, the viscosity, the white color, solubility in water, the ability to not ignite, and the low water loss rate remained stable. We performed the physicochemical characterizations of the obtained materials by using several techniques: X-ray diffraction (XRD), Fourier transform infrared (FTIR), scanning electron microscopy (SEM), transmission electron microscopy (TEM), atomic force microscopy (AFM), thermogravimetric analysis (TGA); and humidity rate determination. We took the XRD measurements in a Bruker D8 Discover diffractometer and used a tube with a copper anode coupled to a Johansson monochromator for Kα, operating at 40 kV and 40 mA in the 20 interval ranging from 5° to 80° for carrying out the analysis. The FTIR analyses of the materials were performed in KBr (3 wt% of the sample contents) in the region from 4000 cm⁻¹ to 400 cm⁻¹, with 16 scans and 4 cm⁻¹ resolution, using a PerkinElmer equipment, model FTIR Spectrum 100. After the sample was fixed on a brass sample holder and covered with a thin gold film, we obtained SEM images to make the material's surface conductive. The images were acquired with a JEOL microscopy, model JSM-IT300, in high vacuum with the secondary electrons detector and electron acceleration voltage ranging from 5 to 15 kV. We carried out TEM analyzes in JEOL equipment, model JEM-2100, with a voltage of 200 kV. We dispersed the samples in hexane for 15 min and then subjected them to a low-frequency ultrasound bath. An aliquot of the obtained dispersion was dripped onto the copper grid covered with carbon and dried. We obtained AFM images with NTMDT AFM equipment in intermittent contact mode. Three different areas of the samples were analyzed. We performed TGA analysis using thermogravimetric equipment from TA Instruments, model SDT Q600 V20.9 Build 20, using an inert Argon atmosphere, with a 100 mL/min flow rate, a heating ratio of 10 °C/min, and temperature ranging from 10 to 1000 °C. The hydrogel’s maximum hydration capacity and its rehydration performance were studied using a thermobalance Bel ITermo instrument, model 163L, with a fixed temperature at 65 °C and room temperature at 24 °C, respectively.

2.5. Dehydration and rehydration of PMF-Si hydrogel

The tests of hydrogel dehydration and rehydration were determined using a digital thermobalance. The dehydration tests were carried out with the sample heating at 65 °C for 5 h. The initial hydrogel weight (water + PMF-Si composite) used to calculate the dehydration percentage was ~10 g in all tests. We performed the rehydration experiments from samples dehydrated from 10 g of hydrogel to lose 70, 75, 80, 85, and 90% in mass at 65 °C. After reaching these mass-loss percentages, we hydrated the samples with the same mass of water lost. The system was then heated to remove the excess of water up to see just the hydrogel phase. We used the final mass of the heated system to calculate the percentage of rehydration. All experiments were performed in triplicate.

2.6. MB removal

We evaluated the adsorption of MB (Vetec) on PMF-Si composite by batches containing 1 g of hydrogel (72 mg of dry PMF-Si) in contact with 10 mL (10-1500 mg/L) of MB solution under stirring at 4000 rpm, for 10 min, at 25 °C58-60. We performed contact time, pH variation, and hydrogel reuse tests at the MB concentration of 50 mg/L. The concentration MB was determined from the absorbance values at 664 nm, using a Genesys UV-Vis spectrophotometer model 10S. We analyzed the solutions at different exposure times ranging from 5 to 120 min to elucidate MB’s kinetic adsorption response on the hydrogel. The data was used to perform the adsorption isotherm where it was determined the adsorption capacity, qe (mg/g), indicating the amount of adsorbed MB dye per amount of dry adsorbent using Equation 3:

\[ q_e = \frac{V(C_0 - C_e)}{m} \]  

where \( V \) is the solution volume (L), \( C_0 \) is the initial MB concentration (mg/L), \( C_e \) is the equilibrium concentration (mg/L) of MB, and \( m \) the adsorbent mass (g).

The potential of zero charge (PZC) of the composite PMF-Si was determined using the HORIBA SZ-100 equipment. Seven suspensions (1% w/v) were produced in 0.01 mol/L KCl solution, varying the pH from 1-7. We used HCl and NaOH solutions to control the average pH. The suspensions were stirred for 12 h, maintained for another 12 h at 25 °C, and then PZC measurements were made with the supernatant. We evaluated the hydrogel’s adsorption capacity for MB removal by varying the pH between 1 and 12. The tests were performed with 1 g of the hydrogel hydrated, 10 mL of MB at a concentration of 50 mg/L, and a temperature of 25 °C. All experiences were done in triplicate.

2.7. Evaluation of hydrogel reuse

We performed the hydrogel reuse tests on MB dye adsorption using 1 g of hydrogel and 10 mL of 50 mg/L MB solution at 25 °C temperature. The dye’s desorption was done by adding 7 mL of 1 mol/L HCl solution, and the contact time was 20 min. After treatment with an acid solution, we performed 4 washing cycles with 12 mL of deionized water. Centrifugation of the solution at 4000 rpm for 10 min was used to remove the acid and water. The MB concentration was determined from absorbance values at 664 nm using a Genesys UV-Vis model 10S spectrophotometer.

3. Results and Discussion

3.1. Fourier transform infrared spectroscopy

Figure 2 shows the FTIR spectra for melamine, PMF, and PMF-Si materials. The infrared spectrum of melamine is presented in Figure 2a, and the founded bands are similar to the one found by Ali et al.61. Bands at 3470 cm⁻¹ and 3419 cm⁻¹ are attributed to the asymmetric stretching of the NH2 group, and two large bands at 3333 cm⁻¹ and 3132 cm⁻¹ are attributed to the symmetric elongation and vibrational bending of primary amine66,65. The bands at 1655, 1555, 1469, and 1439 cm⁻¹ belong to the stretching of C=O bonds.
present in the ring of 1,3,5-triazine of melamine. The band at 814 cm$^{-1}$ is assigned to the vibrational bending of the ring in 1,3,5-triazine of melamine. The band located at 1198 cm$^{-1}$ is assigned to the C-N bond stretching at primary amine. PMF, Figure 2b, showed that the primary amine present in melamine has disappeared. The formation of broadband at 3336 cm$^{-1}$ is attributed to hydroxyl groups' stretching vibration and demonstrates that melamine and formaldehyde are reticulated in the PMF resin. The band around 2956 cm$^{-1}$ is assigned to the stretch of primary carbon bonds, which is absent in neat melamine and occurs in PMF due to the reaction between melamine and formaldehyde. We observed the methylene stretches (CH$_2$) at 1489 cm$^{-1}$. The band located at 1155 cm$^{-1}$ is assigned to the stretching of CO bonds and the band at 1061 cm$^{-1}$ is attributed to an effect of the stretching vibration of ether groups. Bands localized at 1565, 1489, and 810 cm$^{-1}$ are attributed to the melamine rings' vibration. The band at 1330 cm$^{-1}$ derives from the symmetrical stretch deformation of CN bonds. Figure 2c shows the spectrum of the PMF-Si composite. The intense and wideband at 3340 cm$^{-1}$ is due to the stretch vibration of OH of silanol groups in the PMF-Si composite structure, and possibly also due to the presence of residual water, being a consequence of its high hydrophilic character.

Furthermore, the composite has free and encapsulated silanol groups due to the incorporation of Si-NPs into the PMF, Figure 1b. The band at 2943 cm$^{-1}$ is assigned to the symmetrical stretch of the CH bond. The band at 1565 cm$^{-1}$ founded in the PMF is attributed to the vibration of melamine rings, and its displacement to 1571 cm$^{-1}$ in the PMF-Si spectrum reveals an interaction between silanol groups, from silica, and rings of melamine structure. The band at 1050 cm$^{-1}$ results from the C-O stretch of glycol, demonstrating OH groups’ presence stemming from glycerin. The band at 924 cm$^{-1}$ corresponds to the asymmetric stretching of Si-OH groups. The bands at 1111 and 786 cm$^{-1}$ indicate symmetrical extension, while the band at 471 cm$^{-1}$ indicates asymmetrical stretching of Si-O-Si and O-Si-O groups, respectively. Those groups are present in silica belonging to the PMF-Si composite. Those groups account for hydrophilic property, which contributes to water retention by the hydrogel, demonstrating the successful formation of the PMF-Si composite.

3.2. XRD of melamine, SiO$_2$, PMF, and PMF-Si composite

Figure 3 shows diffractograms acquired for melamine, SiO$_2$, PMF, and PMF-Si composite in the 2θ interval ranging from 6 to 80°. Melamine displayed a well-defined crystalline profile with peaks between 15° and 45°. The band at 1155 cm$^{-1}$ shows the spectrum of the PMF-Si composite. The intense and wideband at 3340 cm$^{-1}$ is due to the stretch vibration of OH of silanol groups in the PMF-Si composite structure, and possibly also due to the presence of residual water, being a consequence of its high hydrophilic character.

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3.3. Microscopy analysis

We performed a microscopy examination to acquire SEM, TEM, and AFM images to get an insight into the morphological structure of the synthesized materials. Figure 4 exhibits SEM images of neat melamine, silica, PMF, and
dry PMF-Si composite. Figure 4a shows the melamine morphology consisting of white-color crystalline agglomerates in solid-state at 25 °C. Figure 4b shows the purified silica precipitates presenting a cauliflower-like morphological appearance, representing an amorphous material, and agrees with the XRD spectrum shown in Figure 3. The amorphous silica profile was also verified in other studies.

Figure 4c shows the PMF morphology with a sponge-like porous structure. This arrangement is like the PMF images obtained by Schwarz and Weber. In opposition, Figure 4d demonstrates that the dry PMF-Si composite presents a distinct structure compared to its precursors and resembling dehydrated human skin.

We obtained TEM images for silica and dry PMF-Si composite, Figure 5. Figure 5a shows the silica microparticles with a diameter of about ~0.45 µm. The silica microparticles are interconnected and forming an elongated agglomerate due to its high active surface and elevated density of silanol groups, which is typical of amorphous silica material in agreement with the results already examined. Figure 5b displays the Si-NPs homogeneously distributed within the PMF matrix. The presence of no agglomerated and well-distributed Si-NPs within the PMF-Si composite contributes to achieving better stability and water absorption capability by the hydrogel. From Figure 5b, we determined the Si-NPs diameter using the software ImageJ. These Si-NPs exhibited an average diameter between 10 to 15 nm, as shown in Figure 6. The used precursor concentration of the Si-NPs avoided particle agglomeration, being a positive result for the PMF-Si composite.

Figures 5c and d display the PMF-Si composite morphology. In the lower-left part of Figure 5c, fibers are spaced in the dry composite where large amounts of water can be confined by hydration. In the central region of the same image, fibers are close to each other, featuring an area where a large amount of water was removed from the composite. Figure 5d shows another region in which the PMF-Si composite has a circular geometry, and the fibrous structure is spread out in a 3D network that resembles a sponge structure. In the hydrogel dehydrating process, the PMF-Si composite shows a dense aspect of fibers. Cross-link bonds also characterize PMF. When re-hydrated, the polymer chains pull away, serving as structural support to achieve maximum water absorption. The supplied TEM images demonstrated that the PMF-Si composite consists of fibers and Si-NPs responsible for improved water absorption capacity.

We obtained AFM images of the PMF to better observe the polymer fibers’ morphological aspect, Figure 7. We prepared PMF film by adding 5 mL of the PMF solution on a Petri dish to obtain the images. The sample was heated at...
60 °C in an oven for 24 h to form the polymer film. Then, the film was removed from the plate, cut, and submitted to AFM analysis. The images were acquired in different areas of the film. In these images, we can see the disposal of the fibers in vertical disposal. The fibers are spread all over the film and present an average diameter between 1 and 0.5 μm. Figure 7a shows dendrimers’ formation in the fibers’ terminal positions, which suggest possible separation and forming of new thinner fibers after water absorption forming the hydrogel. The polymer film also shows the fibers juxtaposed, different from what happens when the fibers are inside the hydrogel. In the hydrogel, the polymer fibers are separated to maximize water absorption. The AFM images confirm that the PMF constitutes fibers that promote structural reinforcement to the hydrogel, allowing its stability. Liao et al. also observed the fibrous structure of PMF. Figure 7c is an interfacial approximation between PMF fibers and reveals a phase’s presence, apparently, distant from the fibers, occurring at their edges. We believe that is because the fibers become thicker, characterized as a PMF residue that could make the fiber thicker if the polymerization process were continued. It was not possible to perform the AFM images of the PMF-Si composite. The operational impediment is due to the high hydrophilic nature that does not allow the composite fixation for analysis.

3.4. Thermogravimetric analysis

Figure 8 exhibits the TG and DTG curves of melamine, PMF, SiO₂, and PMF-Si composite. Table 1 summarizes the main events from the thermogravimetric results for the materials mentioned above. Figure 8a shows the melamine thermogravimetric profile with a unique event situated between 220 to 356 °C, without any residue remaining, consistent with the literature, which corresponds to the melamine decomposition. TG and DTG curves of PMF, Figure 8b, exhibited three events related to its thermal properties. The first event occurred between 29 and 325 °C associated with polymer dehydration and decomposition of small precursor molecules of PMF synthesis.

The second event occurred between 362 and 416 °C, demonstrating that the PMF suffered decomposition of less-dense morphological structures such as the residue of
PMF embedded on the surface of the fibers and major of the fibers that presented small diameter, which was previously discussed in Figure 7c. The third event occurred between 420 and 935 °C, attributed to the decomposition of the most compacted PMF structure and thicker fibers. The 10% of residual mass from the PMF decomposition is composed of carbon ashes. The TG and DTG curves of silica, Figure 8c, revealed two relevant events, partly superimposed. The first event located between 48 and 115 °C is attributed to the silica dehydration, followed by a second event related to

![AFM images of PMF film surface at (a) 100 μm², (b) 25 μm², and (c) 9 μm².](image)

![TG and DTG of (a) melamine, (b) PMF, (c) SiO₂, and (d) PMF-Si composite.](image)

| Sample   | °C     | wt% | °C     | wt% | °C     | wt% | °C     | wt% | Residue (%) |
|----------|--------|-----|--------|-----|--------|-----|--------|-----|-------------|
| Melamine | 220-356| 100 |        |     | 362-416| 28  | 420-935| 48  | 10          |
| PMF      | 29-325 | 14  | 362-416| 28  |        |     | 420-935| 48  |             |
| SiO₂     | 48-115 | 8   | 115-660| 5   |        |     |        |     | 87          |
| PMF-Si   | 43-126 | 10  | 126-282| 20  | 282-418| 17  | 418-525| 11  | 42          |
the thermal decomposition of residual organic compounds, e.g., acetic acid, used in the silica synthesis, which extends to ~660 °C. It is possible to verify a high residual content (~87%) characterizing silica content free of water and organic matter. Figure 8d presents the TG and DTG curves of the dry hydrogel containing PMF-Si composite. The curve reveals the occurrence of four relevant events associated with composite dehydration and its thermal decomposition. The first one is attributed to the dehydration process located between 43 to 126 ºC with a mass-loss percentile of ~10%. Although the hydrogel was previously dehydrated at 60 °C for 24 h, its highly hygroscopic nature allowed water absorption, even using a dehumidified storage protocol. The second event between 126 and 282 ºC had a mass-loss percentile of ~20% and corresponds to the organic compounds from silica synthesis. The third event occurred between 282 and 418 ºC, attributed to the glycerin evaporation, which has a boiling point of 290 ºC and presented a mass-loss percentile of about ~17%. The fourth and last event occurred between 418 and 525 ºC, attributed to the PMF fibers’ thermal decomposition, which presented a mass-loss percentile of ~11%. The residue from the PMF-Si composite decomposition is due to the Si-NPs and the carbonized organic matter corresponding to ~42%.

From the results revealed here, it is possible to conclude that the PMF-Si composite presents a compatible thermal behavior with its composition, showing a series of thermal events associated with the dehydrated hydrogel’s content. TG and DTG analyzes indicate that the PMF-Si composite is less stable than the PMF. That is because the fibers in the PMF-Si composite allowed a better material dispersion and, possibly, due to the segregation of thick fibers into thinner fibers while the composite is hydrated. Besides, the presence of Si-NPs favored the dispersion of the PMF fibers benefiting hydrogel stability.

3.5. Dehydration and rehydration results

The dehydration tests showed substantial water loss in the first two hours of heating at 65 ºC representing a decrease in water content of approximately 80%, as shown in Figure 9a. The hydrogel dehydration reached ~0.728 g after 4 h of heating and remained constant for an extra 1 h. The rehydration performance seriously affects the hydrogel’s properties, a critical issue in its practical applications. From Figure 9b, the hydrogel achieves complete rehydration even after reaching 70% of dehydration. Hydrogel dehydration below 70% also showed complete rehydration. We found that weight loss of up to 70% characterizes the hydrogel’s reversibility returning to its original water contents. When we increased the hydrogel dehydration to 90%, its rehydration capacity considerably decreases, such that when dehydrating the hydrogel by 90%, rehydration is about ~30%. The success in hydrogel rehydration (maximum dehydration is up to 70%) is due to its excellent hydrophilic properties. Rehydration is possible because the silanol groups (2 ≡ Si−OH → ≡ Si − O−Si ≡ + H₂O) and hydroxyl groups (Figure 1) are present on the PMF-Si composite surface and between its mesopores structure (Figure 4c), allowing the hydrogel to recover most of the original volume of water through the return effect. The practical consequence of the facile hydrogel rehydration is attributed to the stabler Si-NPs adsorbed on the PMF structure.

3.6. Potential of zero charge and methylene blue removal

The Potential of zero charge (PZC) of the composite occurred at pH = 1.22, as shown in Figure 10a. At MB solution and pH below the PZC, the composite surface is positively charged, and at pH higher than the PZC, the composite surface is negatively charged. Therefore, there is a reduced MB adsorption capacity when the pH is lower than PZC, and this is due to the composite having a positive potential on its surface when the pH is lower than 1.22 (see Figure 10b). However, when the pH is higher than 1.22, the PMF-Si composite surface becomes negative and electrostatic attraction occurs easier with the cationic MB dye molecules. The adsorption by hydrogen interaction justifies that even at lower pH than PZC, the PMF-Si shows an adsorption process. MB adsorption mechanism on Si-NPs is by electrostatic forces and hydrogen interactions on the silane groups since MB is positively charged and has amine groups.

Figure 11 shows a scheme of how adsorption of MB on silane groups can occur. The model shows the interfacial
interaction arising from the silanolate groups’ interactions (Si-O-) derived from silicate, and silane groups, with MB dye because of hydrogen interactions.

The adsorption of methylene blue dye is affected by pH variation. The MB in aqueous solutions is presented in the form of positively charged ions (cationic dye), and its adsorption degree is influenced by the surface charge of the adsorbent, which has its value modified according to the pH variation of the medium. The pH can cause functional groups’ desorption on the adsorbent’s active sites and changes in the surface charge of the adsorbent. The adsorption capacity of MB on PMF-Si increased at pH higher than 2, agreeing with PZC, Figure 10a, and reached stability in the pH range between 4-9. At a pH higher than 9, there is an increase in the adsorption capacity of MB.

The calibration curve from the samples containing different MB dye concentrations in the hydrogel was acquired, Figure 12a. A linear correlation between the absorbance of the samples and the residual dye concentration in the hydrogel is represented by the equation $y = 0.1844x + 0.0078$, with an $R^2 = 0.99946$. The accurate adjustment allowed us to calculate the final MB dye concentration of a different solution using the equation mentioned above. Figure 12b shows the contact time between the MB dye and PMF-Si hydrogel and displays the associated dye removal percentage. The curve indicates the maximum MB dye removal percentage reached 96% with a short contact time of 5 min. Figure 12b also suggests a rapid interaction between the MB dye and the hydrogel. Although the percentage of dye removal continues to increase with the contact time, its increase is not significant, although it allows for additional removal with a contact time of 90 min. Therefore, we achieved a short adsorption time of the dye removal by PMF-Si, which is a relevant result. The high dye removal percentage with a short adsorption time makes MB dye removal by PMF-Si hydrogel a versatile method for practical applications.

Langmuir model was used to fit the absorption data analysis of MB dye in hydrogel samples. This model comprises a
defined number of active sites and the occurrence of monolayer adsorption on a homogeneous surface\textsuperscript{89,90}. The isothermal curve showed in Figure 13 follows the Langmuir adsorption model. The curve’s shape indicates no competition between the solvent and the adsorbate, which means the PMF-Si hydrogel is a potential adsorbent of MB dye.

We calculated the adsorption capacity using Equation 4:

\[
q_e = \frac{q_{\text{max}} K_c C_e}{1 + K_c C_e}
\]

where \(q_{\text{max}}\) represents the theoretical maximum adsorption capacity in mg/g, \(C_e\) is the concentration at the equilibrium in mg/L, and \(K_c\) is the Langmuir’s constant in L/mg\textsuperscript{91}. The adsorption curve (Figure 13) reveals that the hydrogel has a maximum MB dye adsorption capacity of around 140 mg/g and a Langmuir’s constant of about \(3.16 \times 10^3\) L/mg. The calculated MB dye adsorption capacity value is higher than the ones reported for other materials such as rice hulls (~40 mg/g)\textsuperscript{92}, clay (~72 mg/g)\textsuperscript{93}, waste activated carbon (~15 mg/g)\textsuperscript{81}, MCM-41/chondroitin sulfate hybrid hydrogels (~123 mg/g)\textsuperscript{94}, polyvinyl alcohol/acrylic acid/poly4styrene sulphonylic acid hydrogel (~131 mg/g)\textsuperscript{95}. The PMF-Si hydrogel synthesized by Schwarz and Weber used commercial Si-NPs and exhibited an MB dye adsorption capacity of ~812 mg/g\textsuperscript{13}. In our study, Si-NPs were formed in the hydrogel synthesis process, while Schwarz and Weber used commercial Si-NPs. The hydroxylated surface of the PMF-Si composite provided by silanol groups accounted for the strong interaction between the hydrogel and MB dye molecules. Therefore, PMF-Si hydrogel for MB dye removal should be encouraged since it is an effective and inexpensive strategy.

3.7. Evaluation of the reuse of hydrogel for MB adsorption

Figure 14 displays the adsorption capacity of the hydrogel after successive reuse cycles. The process of removing MB from the PMF-Si hydrogel’s surface, in its reuse, occurs by adding 1 mol/L HCl solution, followed by washes with deionized water until pH = 6 is reached. The strategy of lowering the pH to remove MB from the composite surface is because, at pH values below the PZC, there is desorption of many dye molecules. The adsorption capacity of PMF-Si gradually decreased from 92% to 42% after 7 adsorption cycles. The washing process justifies the decrease in adsorption capacity with the HCl solution. The low pH promotes a reduction in the silica surface’s negative charge density,
making the material less MB adsorbent. Another factor to consider is the loss of Si-NPs during successive washing cycles. Therefore, the indication of PMF-Si hydrogel’s use for MB removal is a viable alternative since the materials have a low-cost, and the synthesis is simplified.

4. Conclusions

We successfully synthesized PMF-Si composite to form a hydrogel with promising properties for several applications (dye adsorbent, agriculture, fire-retardant, etc.). Thermal analysis implied that the PMF-Si composite presents a stable structure with well-defined stages of decomposition. The Si-NPs were obtained from purified sodium silicate and exhibited similar properties to the commercial Si-NPs, making this study an alternative for preparing quality and low-cost Si-NPs for hydrogel fabrication. The synthesized PMF-Si composite displayed fibrous structure provided by PMF, responsible for the hydrogel structural support. The Si-NPs showed a diameter between 10 and 15 nm helped to improve the water absorption capacity. The intrinsic PMF-Si composite hydration reached a maximum of 93 wt% of water. Complete rehydration is only possible when the maximum mass loss of water reaches 70 wt%. The composite showed PZC at pH = 1.22, which means that its structure at pH higher than 1.22 is negatively charged, and electrostatic forces occur with the cationic MB dye molecules. The MB dye adsorption capacity, q_max = 140 mg/g, was higher than other materials routinely used for MB dye removal, demonstrating that the PMF-Si hydrogel can be used to remove MB dye and possibly other dyes. Although the adsorption capacity of PMF-Si gradually decreased from 92% to 42% after 7 adsorption cycles, the composite is low-cost, which makes viable its use for MB adsorption proposal.

5. Acknowledgments

This study was financed in parts by the Coordenação de Aperfeiçoamento de Pessoal de Nível Superior –Brasil (CAPES) –Finance Code 001.

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