Spiral Growth and Step Edge Barriers

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Abstract

The growth of spiral mounds containing a screw dislocation is compared to the growth of wedding cakes by two-dimensional nucleation. Using phase field simulations and homoepitaxial growth experiments on the Pt(111) surface we show that both structures attain the same characteristic large scale shape when a significant step edge barrier suppresses interlayer transport. The higher vertical growth rate observed for the spiral mounds on Pt(111) reflects the different incorporation mechanisms for atoms in the top region and can be formally represented by an enhanced apparent step edge barrier.

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When a screw dislocation intersects a crystal surface, a step forms which emerges from the intersection point and cannot be removed by growth or evaporation. Under deposition the step turns around the intersection point, resulting in a characteristic spiral mound. This mechanism was conjectured by Frank in 1949 to account for the enormous discrepancy between experimentally observed crystal growth rates and the predictions of two-dimensional nucleation theory [1, 2]. Since then, spiral mounds have been recognized as a ubiquitous feature of many growth systems ranging from high temperature superconductors [3] to biominerals [4] and organic thin films [5].

Theoretically, the prediction of the shape of a growth spiral involves a moving boundary value problem for the adatom concentration field in a complex geometry [6], which has been treated by approximate analytical calculations [7, 8] as well as by numerical phase field techniques [9]. These approaches result in a simple conical mound shape with a constant step spacing, which is proportional to the step radius of curvature near the spiral core and generally decreases as a power law with increasing deposition flux.

In the growth of organic thin films frequently spiral mounds are observed whose shapes are inconsistent with this classical scenario. They possess small flat plateaus at the top, a slope (or step spacing) that varies non-monotonically with the distance from the spiral core, and they are typically separated by deep grooves [5, 10]. These shapes are strikingly reminiscent of the mound structures, often referred to as wedding cakes, that form during growth on dislocation-free metal surfaces [11, 12, 13, 14] when the mass transport between different layers is suppressed by a strong step-edge barrier. It is then plausible to hypothesize that the unconventional spiral mounds observed on organic thin films may be related to a step-edge barrier, which has been conjectured to exist on the basis of various indirect pieces of evidence [5, 15, 16, 17].

In the present work we therefore investigate experimentally and theoretically the hitherto disregarded situation of spiral growth in the presence of a step edge barrier. Our starting point is the observation that an atom finding itself on the hillside of a spiral mound, far away from the core, diffuses in an environment that is indistinguishable from that on the hillside of a wedding cake consisting of concentric circular islands [7]. The overall shape of the spiral mound must therefore reflect the strength of the step edge barrier in the same way as for a wedding cake.

This is illustrated in Fig. 1 for a growth spiral generated numerically using a phase field
FIG. 1: (Color online) Growth spirals obtained from phase field simulations with $D = 10$, $F = 0.2$, $\gamma = 1$, $\rho^* = 0.1$, $k_+ = 10$, $k_- = 1$. Panels show the surface before growth (left) and after deposition of 140 ML (middle) and 300 ML (right), respectively.

model [18, 19]. The model is a diffuse interface approximation of the Burton-Cabrera-Frank (BCF) theory [2], which takes the form of a moving boundary value problem for the adatom concentration on the surface. Key parameters of the model are the surface diffusion coefficient $D$ and the deposition flux $F$, as well as the kinetic coefficients $k_\pm$ governing the attachment of atoms to a step from the lower ($k_+$) and upper ($k_-)$ terrace, respectively. Step thermodynamics enters through the step stiffness $\gamma$ and the equilibrium adatom concentration $\rho^*$. In the phase field approximation the crystal surface is represented by a continuous field $\phi(\vec{r}, t)$ which couples to the adatom concentration and is subjected to a multiwell potential that pins $\phi$ to the preferred integer values representing the discrete surface heights. The step edge barrier is incorporated through a mobility function that is minimal at the steps and asymmetric with respect to half-integer values of $\phi$. In the numerical implementation the model is non-dimensionalized by scaling lengths and times with suitable microscopic quantities.

A screw dislocation is introduced by the shift $\phi \to \phi - \theta(\vec{r})/2\pi$, where $\theta(\vec{r})$ is the planar rotation angle around the dislocation core [9, 19]. In the absence of a step edge barrier the simulation of the model converges to a conical mound with a spatially constant, time-independent slope. By contrast, the spiral mound displayed in Fig. 1 has a flat top and curved hillsides which steepen indefinitely with increasing deposition flux, up to the point where further steepening is limited by the intrinsic step width of the phase field. This is
precisely the behavior expected from the analogy with wedding cakes [11, 14].

Despite the similarity in the large scale shape, it is clear that the growth mechanism near the top of the structure is completely different for spiral mounds and for wedding cakes. Even the answer to the simple question as to which of the two grows faster under a given set of conditions is not obvious. The growth of wedding cakes is governed by the rate of two-dimensional nucleation of islands on the top terrace, which is enhanced by the confinement of atoms due to the step edge barrier [20]. In contrast, atoms landing near the top of the spiral mound can avoid the step edge barrier by moving around the core, which should reduce the local adatom concentration and hence the growth rate. On the other hand, the indelible presence of the step emanating from the screw dislocation obviates the need for two-dimensional nucleation, which should increase the rate of vertical growth. When the adatom concentration in our simulations was allowed to become sufficiently large so that islands nucleated deterministically, spiral mounds were generally seen to grow higher than regular wedding cakes [19], indicating that the second effect dominates.

To experimentally compare the two growth mechanisms under identical conditions, we have devised a setup in which they are simultaneously realized on the Pt(111) surface. The experiments were performed in an ultra high vacuum variable temperature scanning tunneling microscopy (STM) apparatus with a base pressure in the $10^{-11}$ mbar range [21]. Cleaning of the Pt(111) sample was accomplished by cycles of flash annealing to 1273 K and sputtering by a mass separated 5 keV Ar$^+$ ion beam. In order to fabricate a surface with screw dislocation lines intersecting the surface in a suitable areal density, prior to growth the procedure outlined in [22] was applied. Helium was implanted at room temperature into the sample by a 4.5 keV He$^+$ ion beam incident at an angle of 76° with respect to the surface normal and with a dose of $3.8 \times 10^{20}$ ions/m$^2$. Subsequently the surface was annealed for 120 s to $T = 800$ K. The growth of the precipitating He bubbles gives rise to dislocation loop punching. Dislocation reactions and glide towards the surface results partially in loop annealing (adatom island formation) but partially also in separated pairs of screw dislocation intersection points with the surface (step dipoles). Onto this surface Pt was evaporated with a typical rate of $1 \times 10^{-2}$ ML/s, where 1 monolayer (ML) denotes the surface atomic density of Pt(111). The deposited amounts $\Theta$ range from 1 to 100 ML and temperatures $T$ from 250 to 500 K. After deposition the morphology was quenched to avoid changes due to surface diffusion.
FIG. 2: (Color online) (a) Morphology of Pt(111) after deposition of 39 ML at 400 K on a substrate area with screw dislocation intersection points. Image size 1750 Å × 1750 Å. Inset: Enlarged view of the top area of the spiral mound labeled A. (b) Radially averaged height distributions of the spiral mound labeled A in (a) [full red/gray line] and the wedding cake labeled B in (a) [full black line]. The fits to the shape profiles according to the analytic model are plotted as dashed lines.
Figure 2 (a) represents the morphology resulting after deposition of a total coverage Θ of 39 ML at 400 K. As in previous experiments on dislocation-free surfaces [13, 14], a large number of mounds are formed that exhibit flat tops and are separated by deep grooves. Two of the mounds visible in Fig. 2 (a) are significantly higher than all others; their heights exceed the surrounding mounds by 4-6 ML. Closer inspection reveals that both of them carry a screw dislocation intersection point in their center, as exemplified by the inset. Figure 2 (b) displays the mound profiles obtained from the height distributions of the spiral mound labeled A and the wedding cake labeled B in Fig. 2 (a). The profiles are obtained by redistributing the layer coverages into a stack of circular islands. Both mounds clearly show flat top plateaus. The spiral mound profile displays in addition a change from positive curvature close to the top to negative curvature close to the edge. Such an inflection point in the height profile is characteristic for mounds grown in the presence of large step edge barriers [14].

The dashed lines in Fig. 2 (b) represent fits to the function $\theta(h) = 1 - C\{1 + \text{erf}((h - \Theta)/\sqrt{\Theta})\}$, where $\theta(h)$ is the fractional coverage at height $h$ and erf denotes the error function. The width of the mound at height $h$ is proportional to $\sqrt{\theta(h)}$. This equation follows from a simple model for the growth of wedding cakes [23], in which mass transport between all layers except the top one is completely suppressed and a new top terrace nucleates when the current one has reached a critical coverage $\theta_c$ [14]. The only fit parameter $C$ is related to $\theta_c$ and is most simply determined from the total height of the mound. While the agreement with the experimental data is not perfect, the fit is satisfactory in view of the crudeness of the model and the large number of factors that have been neglected in its application. For example, repulsive step-step interactions come into play when the steps on the hillside approach each other and tend to reduce the mound slope, as observed in Fig. 2 (b).

Via the theory of second layer nucleation [14, 20], the critical coverage $\theta_c$ can be linked to the additional step edge barrier $\Delta E_S$, which is defined as the difference between the activation energy for a descending hop into the step position compared to a hop within the layer [24]. A large value of $\Delta E_S$ enhances the confinement of adatoms on the top terrace, which increases the probability for nucleation, decreases $\theta_c$, and increases the height of the mound. Using the known data for adatom diffusion on Pt(111) [14], the fits shown in Fig. 2 (b) translate into the estimates $\Delta E_S = 0.22$ eV for the wedding cake (mound B) and $\Delta E_S = 0.34$ eV for the spiral mound (mound A). Although spiral mounds obviously do not grow by
two-dimensional nucleation, we can formally represent the experimentally observed faster growth of the spiral mounds by an increase of the apparent step edge barrier by 0.12 eV.

Another prediction of the model is that the height difference between two mounds characterized by different values of $C$ or $\theta_c$ grows proportional to $\sqrt{\Theta}$. This prediction should then also apply to the average height difference $\Delta h$ between spiral mounds and wedding cakes. Figure 3 shows experimental data for $\Delta h$ obtained by measuring the extra height of a spiral mound with respect to the heights of the wedding cakes being its direct neighbors. This quantity was then averaged for each temperature $T$ and coverage $\Theta$ over an ensemble of spiral mounds. Indeed Fig. 3 shows that $\Delta h$ grows roughly proportional to $\sqrt{\Theta}$ for the two temperatures analyzed.

As a further test of the concept of an increase in the apparent step edge barrier for spiral mounds, we have determined the sizes of the flat top region for mounds grown at various temperatures with a total coverage of 10 ML [see Figs. 4(a)-(c)]. To characterize the extension of the top area for spiral mounds, the sum of the lengths of step segments 7 to 12

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FIG. 3: Average height difference $\Delta h$ between spiral mounds and wedding cakes for 400 K (full circles) and 450 K (full squares) as a function of $\Theta$. The dashed lines correspond to $\Delta h \propto \sqrt{\Theta}$ as predicted by the mound growth model.
FIG. 4: (Color online) Examples of spiral mounds after deposition of 10 ML at (a) 300 K, (b) 400 K, (c) 500 K. Image size is always 400 Å × 400 Å. Schematic top view sketch of the top area of (d) a wedding cake and (e) a spiral mound. The numbers label the step segments. (f) Results for $\Delta E_S$ for wedding cakes (full dots) for the three growth temperatures investigated together with the apparent $\Delta E_S$ for spiral mounds (full squares) grown simultaneously.

of the spiral is used as sketched in Figs. (d)-(e). This corresponds to the perimeter of the base terrace, the terrace below the top terrace of a wedding cake, which is known to show less scatter than the top terrace size itself [20]. Using formulae of second layer nucleation theory [14], the base terrace perimeter translates into estimates for $\Delta E_S$ for the two kinds of mounds, which are displayed in Fig. (f). We find a consistent increase of the apparent step edge barrier of spiral mounds by 0.11 - 0.13 eV, in good agreement with the result obtained
from the mound shapes in Fig. 2.

The results presented so far raise an obvious question: Is the faster growth of spiral mounds observed on Pt(111) a generic feature, or is it specific to this growth system? To provide at least a qualitative answer, note first that only the fate of the adatoms arriving in the top region [the areas shaded light gray in Figs. 4(d) and (e)] determines the shapes and heights of mounds, given that $\Delta E_S$ is significant, because then adatoms arriving on the slopes of a mound are incorporated with a probability near unity into the ascending step bounding their terrace. In contrast, adatoms arriving near the top of a mound are able to reach a lower layer, either because the top terrace on which they land is small (in the case of a wedding cake) or because they can move around the spiral core. Thus the area of the top region is directly proportional to the effective downhill flux of adatoms. The smaller the top region, the higher the mound and the closer the mound shape is to the statistical growth limit, in which the layer coverages have a Poisson distribution [11, 14].

Understanding the difference between the two kinds of mounds therefore amounts to understanding the physics that determines the size of the top area. For wedding cakes the key process is two-dimensional nucleation [14, 20]. For the spiral mounds on Pt(111), the appropriate physical picture is that of a polygonized spiral growing by the motion of straight segments that follow the crystallographic directions of the surface [6]. The scale of the top region is then set by the length $l_c$ that the stationary segment 1 in Fig. 4(e) has to reach before it can grow by incorporating adatoms, thus taking over the role of segment 2. Hence $l_c$ is the critical segment length for one-dimensional nucleation, and it is determined by the atomistic processes of step edge diffusion and corner rounding [14]. Applying the kinematic approach of [6] to the fcc(111) surface, one finds that $l_c$ equals the length of segment 2, which does not change in time. In the present case this yields the estimate $l_c \approx 23\pm6\text{Å}$, which is much smaller than the average top terrace size of a wedding cake. This discussion makes it clear that for a system with a different hierarchy of diffusion processes the effect of a spiral step might be qualitatively different. If the activation energy for step edge diffusion is small and the step edge barrier is large, $l_c$ could be large compared to the size of a top terrace at second layer nucleation, and wedding cakes could be higher than spiral mounds. In fact, this has to be the case in the limit of an infinite step edge barrier, $\Delta E_S \to \infty$, where the top terrace of a wedding cake shrinks to a point [11] while the spiral mounds maintain a finite top plateau of size $\sim l_c$. 
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[23] The equation applies in the limit of large $\Theta$, where the fact that the height distribution of a wedding cake is discrete, in contrast to the continuous distribution of a spiral mound, is irrelevant.

[24] We use $\Delta E_S$ as effective barrier averaging over the atomistic details of the step configuration, which in fact give rise to a variation of the barrier with step atomic structure and also neglect details of the potential energy landscape at the step edge (compare [14]).