All-electron quasi-particle self-consistent $GW$ band structures for SrTiO$_3$ including lattice polarization corrections in different phases

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The electronic band structure of SrTiO$_3$ is investigated in the all-electron QS$GW$ approximation. Unlike previous pseudopotential based QS$GW$ or single-shot $G_0W_0$ calculations, the gap is found to be significantly overestimated compared to experiment. After putting in a correction for the underestimate of the screening by the random phase approximation in terms of a $0.8\Sigma$ approach, the gap is still overestimated. The $0.8\Sigma$ approach is discussed and justified in terms of various recent literature results including electron-hole corrections. Adding a lattice polarization correction (LPC) in the $q \rightarrow 0$ limit for the screening of $W$, agreement with experiment is recovered. The LPC is alternatively estimated using a polaron model. We apply our approach to the cubic and tetragonal phases as well as a hypothetical layered post-perovskite structure and find that the LDA (local density approximation) to $GW$ gap correction is almost independent of structure.

I. INTRODUCTION

It is well known that the density functional theory in its commonly used local density and generalized gradient approximations (LDA and GGA) does not provide accurate electronic band structures and in particular underestimates band gaps. This is by now recognized to be mostly because the Kohn-Sham eigenvalues in this theory should not be interpreted as one-electron excitations. To calculate the latter, a many-body-perturbation theory, including a dynamical self-energy, such as the $GW$ approximation, provide a much better justified and more accurate framework. For standard tetrahedral semiconductors, the $GW$ method has been shown to provide accurate gaps. Still, this depends on details of the implementation, for example, all-electron results may differ from pseudopotential results and the level of self-consistency used in the $GW$ method and its convergence versus various parameters plays a significant role. For transition metal and complex oxides, it is still far less clear how well the $GW$ method performs. Here we consider SrTiO$_3$ as a case study.

We use the all-electron full-potential linearized muffin-tin orbital (FP-LMTO) implementation of the quasi-particle self-consistent (QS) $GW$ method and compare its results for SrTiO$_3$ with previous results in literature.

II. LITERATURE REVIEW

Sponza et al. performed $G_0W_0$ calculations of the band structure starting from a pseudopotential LDA calculation including Sr $4s, 4p$ and Ti $3s, 3p$ semicore states as valence. They obtain the vertical gap at $\Gamma$ to be 3.76 in good agreement with experiment, whereas their LDA calculation gave 2.21 eV. The actual valence band maximum (VBM) at $R$ is slightly higher than at $\Gamma$ resulting in a smaller indirect gap both in LDA and in $GW$. The focus of their paper is on the optical dielectric function including electron-hole interaction effects.

Hamann and Vanderbilt (HV) performed QS$GW$ calculations using maximally localized Wannier functions (MLWF) to interpolate the self-energy $\Sigma$ matrix between $k$ mesh-points on which the QS$GW$ is performed. A similar functions is played by the atom centered muffin-tin-orbitals in our approach. They include only Sr-$4p$ semi-core states as valence electrons. Both these groups used the ABINIT package but used somewhat different cut-off parameters. Their plane-wave cut-off for the basis set is similar but HV used a smaller number of unoccupied bands. They obtained the indirect LDA gap of 1.61 and a $GW$ gap of 3.32 eV. Curiously, the gap correction of HV (1.71 eV) is larger than that of Sponza et al. (1.55 eV). They did not mention the direct gap at $\Gamma$, but assuming all LDA calculations considered here get similar value for this difference, we’ll use our LDA value (0.44 eV) for the difference between the VBM at $R$ and $\Gamma$. HV’s direct gaps at $\Gamma$ would then amount to 2.05 eV (LDA) and 3.76 eV ($GW$). Thus, these two pseudopotential calculations are in good agreement with each other in spite of the
small changes in parameter choices. The main point of HV’s paper is that the MLWF interpolation works well and indicates little change in the Wannier functions extracted from LDA or GW calculations.

A third pseudopotential based GW calculation by Cappellini et al\textsuperscript{[2]} obtained significantly different results. They also include Sr 4s, 4p, Ti 3s, 3p as valence electrons and obtain an LDA gap at $\Gamma$ of 2.24 eV (indirect $R - \Gamma$ of 1.90) but GW gaps of 5.42 eV ($\Gamma - \Gamma$) and 5.07 eV ($R - \Gamma$). The reason for this discrepancy is unclear but presumably is related to the use of a model dielectric function instead of a consistently calculated one. Finally, a previous FP-LMTO QSGW calculation by Kotani et al. \textsuperscript{[8]} gives the indirect gap at $\Gamma$ of about 4.25 eV but gave few details.

From the above, it appears from the pseudopotential calculations that the $G_0W_0$ gap is close to that of the QSGW gap, and that both are in good agreement with experiment. The all-electron QSGW gap however seems to be about 1 eV larger than experiment. Here we further investigate this issue.

III. METHODS

The QSGW approximation as implemented in FP-LMTO was described in detail in Ref. \textsuperscript{[4]}. The idea behind the QSGW method is to make an optimal choice of the $H_0$ Hamiltonian so that its Kohn-Sham eigenvalues $\epsilon_i$ are as close as possible to the quasiparticle energies $E_i$. To do this, a hermitian but non-local exchange correlation potential, specified by its matrix in the basis of the $H_0$ eigenstates,

$$[V^\Sigma_{xc}]_{ij} = \frac{1}{2} \text{Re}[\Sigma(E_i) + \Sigma(E_j)],$$

is used in $H_0$. Here, $\Sigma(\omega)$ is the energy dependent self-energy calculated from $G_0(\omega)$, the one-electron Green’s function corresponding to $H_0$, in the single-shot GW approximation: $\Sigma = iG_0W_0$. Starting from an LDA $H_0$, $\Sigma$ is calculated, $V^\Sigma_{xc} - V^{LDA}_{xc}$ is added to $H_0$, a new $G_0$ calculated and so on till self-consistency. The reasons behind this approach and differences from fully self-consistent scGW are discussed in Refs. \textsuperscript{[4] \textsuperscript{11} \textsuperscript{12} and 10}.

For tetrahedral semiconductors, this approach provides systematically a $\sim 20\%$ overestimate of the gap due to the underestimate of the dielectric screening in the random phase approximation (RPA) which does not include electron-hole effects and thus misses ladder diagrams in the evaluation of the irreducible polarization propagator $\Pi^0 = -iG_0 \times G_0$, which determines $W$ through $W = (1 - v\Pi^0)^{-1}v$, where $v$ is the bare Coulomb interaction and a simplified symbolic operator notation is used. This has led to the adoption of a universal 0.8$\Sigma$ correction factor \textsuperscript{[13] \textsuperscript{14} \textsuperscript{15} \textsuperscript{16}}. This is illustrated in Fig. 1 which shows the typical underestimate of screening by QSGW to be 20% as indicated by the dashed line. Although it is not clear a priori that this also applies to oxides we adopt a similar correction factor here.

It is interesting that $\epsilon_\infty$ predicted by the LDA is in sometimes better agreement with experiment. This can be attributed to a fortuitous cancellation of errors: missing ladder diagrams tend to cause $\epsilon_\infty$ to be underestimated, while the LDA’s gap underestimate contributes an error of the opposite sign. There is no universal pattern, however, as is already apparent in the data shown in Fig. 1. Where gap errors are severe the LDA severely overestimates the $\epsilon_\infty$. For example in NiO, the $\epsilon_{\infty}^{LDA}$ is 30.

Further justification for the 0.8$\Sigma$ correction factor can be obtained from the work of Shishkin, Marsman and Kresse (SMK)\textsuperscript{[13]} and Wei and Pasquarello (WP)\textsuperscript{[12]} who added an exchange-correlation kernel to the screening of the polarization function $\Pi = [1-(v+fvxc)\Pi^0]^{-1}$ using the nanoquanta kernel or a bootstrap kernel respectively. We will refer to their approach as QSGW. Although these kernels primarily address the $q \to 0$ and static ($\omega = 0$) behavior and might thus not capture the full extent of the electron-hole effects on renormalizing the screening in $W$, and have received some critical discussions,\textsuperscript{[13]} it is useful in the present context to analyze how much they affect the gaps for a variety of materials. Analyzing the data in Table I in WP, part of which is reproduced here in Table 1 with additional analysis, we find that $|E_g(QSGW) - E_g(LDA)|/|E_g(QSGW) - E_g(LDA)|$ has an average value of about 0.76 with standard deviation of 0.04 with the largest deviation for NiO, where it is 0.85 and ZnO, 0.68. ZnO, is a notably difficult material to converge and SMK’s values for the QSGW and QSGW gaps would give 0.77. NiO is a well-known strongly corre-
lateral material and a deviation here is not too unexpected. We note that multiplying the self-energy operator Σ by 0.8 is not exactly the same as correcting the gap shift by 0.8. A slightly larger gap reduction typically occurs.

Very recently, Kutepov’s scheme B, D introduce a way to solve Hedin’s full set of equation beyond the GW approximation using systematic diagrammatic approximations for the vertex function. First of all, his results show fully self-consistent scGW results differ only slightly from the QSGW results and tend to overestimate the band gaps by a similar amount. Secondly, he used two different self-consistency schemes which both introduce vertex corrections both in G and Π. The results of his scheme B, which in his notation only includes a first correction to the vertex Γ₁, are close to those of SMK and WP where comparison for the same material is possible (Si, LiF, GaAs, SiC, BN, MgO) while his most advanced scheme including the full Γ₉₉ vertex, give a somewhat larger gap reduction typically occurs. These are also shown in Table I.

Viewed as percentage of the scGW-GGA (or LDA) correction they give correction factors of about 0.78 and 0.72 respectively when averaged over various cases. As an example for MgO, his scGW gap is 9.31 and his schemes B and D give 8.24, 7.96 eV while WP’s QSGW and QSGW give 9.29, 8.30 eV and SMK obtain 9.16 eV, 8.12 eV respectively. The scheme D agrees almost perfectly with experiment when a lattice-polarization correction of 0.15 eV added to the experimental volume but the latter may be somewhat underestimated. Results also support that the electron-hole correction effects beyond RPA amount to about a 20 % reduction of the QSGW gap correction beyond LDA or GGA.

Besides the electron-hole corrections discussed until now, we also consider a lattice-polarization correction as suggested by Botti and Marques (BM) and revisited recently in Ref. 20. The idea here is that for strongly ionic materials, with large LO-TO phonon splittings, the W in the long-wavelength limit W(q → 0, ω) should include the effects of the ionic displacements on the macroscopic dielectric constant. The macroscopic dielectric constant enters the calculation of Σ in the special treatment of the q → 0 region in the convolution integral over k-space:

\[
\Sigma_{nm}(k, \omega) = \sum_{\mu
u} \int \omega' \sum_{q} \sum_{n'} \sum_{q'} G_{nn'}(k - q, \omega - \omega') W_{\mu\nu}^q(q, \omega') e^{-i q \cdot n'} \langle \psi_{kn} | \psi_{k-qn'} | E_n^q \rangle | \langle E_n^q | \psi_{k-qn'} | \psi_{kn} \rangle \tag{2}
\]

Here, a two-particle mixed product interstitial-plane-wave basis set E₉ diagonalizing the bare Coulomb interaction matrix is used and W is the correlation part of W, subtracting the bare exchange. The need for a special treatment of the q → 0 region arises from the integrable divergence of the Coulomb interaction (\( \propto 1/q^2 \)) and is here treated using the modified offset-Γ method which in turn is closely related to the analytic k · p scheme of Friedrich et al. This involves the macroscopic dielectric

| Material | LDA E₀ | QSGW E₀ | QSGW-LDA E₀ | QSGW-LDA E₀ | Ratio | scGW E₀ | B | D | scGW-LDA E₀ | B-LDA E₀ | D-LDA E₀ | B | D | Ratio | Stdv |
|----------|--------|---------|------------|------------|-------|--------|---|---|------------|--------|--------|---|---|-------|-----|
| MgO      | 4.65   | 9.29    | 8.30       | 4.64       | 3.65  | 0.79   | 9.31| 8.24| 7.96       | 4.66   | 3.59   | 3.31| 0.77| 0.71  |
| NiO      | 1.05   | 4.97    | 4.40       | 3.92       | 3.35  | 0.85   |     |     |            |        |        |     |     |       |     |
| TiO₂     | 1.90   | 4.22    | 3.73       | 2.32       | 1.83  | 0.79   |     |     |            |        |        |     |     |       |     |
| Cu₂O     | 0.53   | 2.65    | 2.12       | 2.12       | 1.59  | 0.75   |     |     |            |        |        |     |     |       |     |
| ZnO      | 0.85   | 4.61    | 3.42       | 3.76       | 2.57  | 0.68   |     |     |            |        |        |     |     |       |     |
| C        | 4.22   | 6.4     | 5.9        | 2.18       | 1.68  | 0.77   | 6.15| 5.8 | 5.73       | 1.93   | 1.58   | 1.51| 0.82| 0.78  |
| SiC      | 1.39   | 2.9     | 2.52       | 1.51       | 1.13  | 0.77   | 2.89| 2.52| 2.42       | 1.50   | 1.13   | 1.03| 0.75| 0.69  |
| GaAs     | 0.43   | 1.75    | 1.51       | 1.32       | 1.08  | 0.75   | 2.27| 1.80| 1.72       | 1.84   | 1.37   | 1.29| 0.74| 0.70  |
| BN       | 4.53   | 7.51    | 6.67       | 2.98       | 2.14  | 0.74   |     |     |            |        |        |     |     |       |     |
| LiCl     | 6.52   | 10.98   | 9.87       | 4.46       | 3.35  | 0.74   |     |     |            |        |        |     |     |       |     |
| Si       | 0.57   | 1.47    | 1.3        | 0.9        | 0.73  | 0.76   | 1.55| 1.32| 1.26       | 0.98   | 0.75   | 0.69| 0.77| 0.70  |
| LiF      | 9.28   | 15.90   | 14.50      | 6.62       | 5.22  | 0.79   | 16.3| 15.02| 14.39      | 7.02   | 5.74   | 5.11| 0.82| 0.73  |
| Ge       | 0.00   | 0.96    | 0.82       | 0.96       | 0.82  | 0.76   |     |     |            |        |        |     |     |       |     |
| AlP      | 1.60   | 3.1     | 2.77       | 1.5        | 1.17  | 0.75   | 2.84| 2.53| 2.44       | 1.24   | 0.93   | 0.84| 0.75| 0.68  |
| CdS      | 1.21   | 3.41    | 2.74       | 2.2        | 1.53  | 0.75   |     |     |            |        |        |     |     |       |     |
| Average  |       |         |            |            | 0.76  |       |     |     |            |        |        |     |     |       |     |
| Stdv     |       |         |            |            | 0.04  |       |     |     |            |        |        |     |     |       |     |

* From SMK[21]
tensor $\mathbf{L}(\omega)$, in their notation $e^T \mathbf{L}(\omega) e_k$. The projection along unit vectors $e_k$ takes care of the non-analytic (orientation dependent) nature of the $k \rightarrow 0$ limit and fully takes into account any possible anisotropies depending on the crystal structure. It is this macroscopic dielectric tensor, usually written $\varepsilon(\omega)$ which needs to be modified to take into account the lattice polarization effect. This is most easily done by means of a Lyddane-Sachs-Teller factor:

$$\frac{\varepsilon^a_{\text{plc}}(q \rightarrow 0, \omega)}{\varepsilon^a_{\text{el}}(q \rightarrow 0, \omega)} = \prod_m \left( \frac{\omega^2_{\text{LO}(m)}}{\omega_{\text{TO}(m)}} - (\omega + i\delta)^2 \right)^{\frac{1}{2}},$$

where the superscript $a$ denotes a projection direction of the tensor, $(\varepsilon^a = \varepsilon^T_a e_a)$. It is clear from this expression that the correction goes to zero for $\omega \ll \omega_L$. In practice we only include it for $\omega = 0$ to avoid the necessity for a careful integration mesh right near the phonon frequency poles.

As discussed in Ref. [20] the BM approach gives the long-range or Fröhlich contribution to the Fan-part of the zero-point motion electron-phonon correction of the gap. The $q$-point integration mesh that needs to be used is a subtle issue discussed in Lambrecht et al. [20]. The strength of this contribution, applied only at $q = 0$ for convenience, can be estimated from the polaron length scale, $a_p = \sqrt{\hbar / 2m_L \omega_L}$ with $m_L$ the band-edge effective mass and $\omega_L$ the relevant LO-phonon frequency. We will discuss later how to apply this in the present case with multiple phonons and a degenerate VBM not occurring at $\Gamma$. The polaronic point of view allows us to make an independent estimate of the corresponding gap reduction.

### IV. COMPUTATIONAL DETAILS

We employ a generalized FP-LMTO method [12] as implemented in the Questaal package [23]. The basis set is specified by two sets of parameters, the smoothing radii $R_{sm}$ and decay lengths ($\kappa$) of smoothed Hankel function envelope functions [23]. For SrTiO$_3$ we include (spd, spd) for Sr, (spd, spd) for Ti and (spd, sp) for O atoms respectively. These indicate the angular momenta included for each $\kappa$. The envelope functions are augmented inside the spheres in terms of solutions of the Schrödinger equation and their energy derivative up to an augmentation cutoff of $l_{\text{max}} = 4$. In addition, calculations are made with and without the 4$p$ (3$p$) local orbitals inside the spheres for Sr and (Ti).

The Brillouin zone integration $k$-point convergence and other convergence parameters of the method were carefully tested for cubic SrTiO$_3$ and similar criteria were adopted for the tetragonal and orthorhombic phases. We also tested result with a larger $k$-point mesh and found the band gap is converged within 0.05 eV. Specifically, we used a $4 \times 4 \times 4$ un-shifted mesh for the Brillouin zone sampling, along with the tetrahedron method for the cubic cases in the LDA self-consistent charge convergence and for the calculation of the $\Sigma$ in GW. For the tetragonal phase, the unit cell is larger along the $c$-direction than in-plane by a factor $\sqrt{2}$. Thus, we use accordingly smaller number of $k$-points, $4 \times 4 \times 3$ for both LDA and QSGW calculations.

For the self-consistency cycle, the charge density and the total energy are converged within the tolerance of $10^{-5}$ $e/a_0^3$ and $10^{-5}$ Ry respectively. For QSGW, after several convergence test calculations, we settled the cut-off above which the self-energy matrix is approximated by an average diagonal value, $\Sigma_{\text{cut}} = 3$ Ry, including self-energy calculations up to 3.5 Ryd, the interstitial plane wave cut-off energy for basis functions $E_{\text{cut}}(\psi_G) = 2.6$ Ry and for the auxiliary basis $E_{\text{cut}}(\psi_{\text{cout}}) = 2.8$ Ry respectively. In QSGW, the self-consistent iteration was carried until the change in $\Sigma$ was less than $10^{-4}$ Ry.

### V. CRYSTAL STRUCTURES

We consider the cubic and tetragonal anti-ferro-electrically distorted (AFD) $I4/mcm$ structure occurring at low temperature. In addition we consider the layered orthorhombic CaIrO$_3$ structure, suggested to occur at high pressures by Cabaret et al. [20] and also known as the post-perovskite structure. Although we will show elsewhere [27] that this structure is unlikely to occur because it has a higher equilibrium lattice volume and much higher total energy, it is of interest to see how the GW gap corrections compare in such different structures. Fig. 2 shows the crystal structures for cubic, tetragonal and orthorhombic from left to right respectively.

Table II summarizes the structural parameters used in the calculations, such as the lattice constants and Wyckoff positions. The relaxed lattice constant for the cubic phase in LDA is 3.86 Å which is only 1% underestimated relative to the experiment.
VI. RESULTS

A. Cubic STO

In Fig. 3a we show the band structure of cubic SrTiO\(_3\) in the full QS\(_0\)GW approach compared with LDA. A few states at \(\Gamma\) are symmetry labeled for later reference. In Table III we summarize the gaps and various other band structure parameters in different approximations. In Table V we show how the different approximations affect other band states relative to the VBM. This allows us to assess to what extent the QS\(_0\)GW correction can be approximated by a \(k\) and state independent scissor shift.

First, we see that our LDA gap agrees quite well with other LDA (or GGA) calculations. Second we see that the QS\(_0\)W\(_0\) gap is significantly lower than the QS\(_0\)GW gap. Third, unlike the pseudopotential calculations reviewed in Sec. II, the QS\(_0\)GW gap significantly overestimates the gap. Even if we use the 0.8\(\Sigma\) approach, they are still larger than experiment. It is only when we add both the 0.8\(\Sigma\) and lattice polarization correction, that we recover the experimental values. We also note that the 0.8\(\Sigma\) approach actually reduces the QS\(_0\)GW-LDA indirect (direct) gap shifts by about a factor 0.73 (0.74). In agreement with other calculations and already correctly described in LDA, the indirect \(R - \Gamma\) gap is about 0.4 eV lower than the lowest direct \(\Gamma - \Gamma\) gap. The VBM between \(R - M\) is very flat and in QSGW the actual VBM lies actually in between \(R\) and \(M\) and is 0.09 eV above that at \(R\). Finally we see that the semi-core levels play a more important role in QSGW than in LDA. Neglecting them, the gap would be only 0.07 eV lower in LDA but is 0.5 eV lower in QS\(_0\)GW or still 0.2 eV lower in the final LST and 0.8\(\Sigma\) corrected case.

B. Polaron estimates

Next, we discuss the lattice polarization correction to the gap in detail. The zero-point motion correction contains a contribution from the long-range Fröhlich type of electron phonon coupling. The latter is arguably the largest electron-phonon coupling correction for a strongly ionic material with large LO-TO splitting because the other electron-phonon coupling effects tend to be smaller than 0.1 eV except for systems with all light atoms. To estimate it we follow the approach of Nery and Allen.\(^{35}\) The main point is that the Fröhlich electron-phonon coupling behaves as \(1/q\) and hence near band edges where the band difference \(E_n(k + q) - E_n(k)\), which enters the denominator in the Allen-Heine-Cardona expression for the electron-phonon self-energy, gives a divergent contribution. Nery and Allen showed how it can be integrated analytically when a simple effective mass approximation is used for the bands. The length scale for the polaron effect is \(a_P = \sqrt{\hbar/2m^*\omega_L}\) and if we assume we need to integrate the singular behavior only over a region in \(q\)-space of size \(1/a_P\) as upper limit, then the polaron shift of a band is given by\(^{20}\)

\[
\Delta E_n(k) = -\alpha_P \hbar \omega_L / 2 \quad \frac{\hbar}{2m^*} \left( \frac{1}{\varepsilon_{\infty}} - \frac{1}{\varepsilon_0} \right),
\]

In other words it essentially the change in the Coulomb interaction calculated at the polaron length scale due to the change in screening from only electronic screening to electron plus lattice screening. The extra factor 2 arises from the choice of cut-off in \(q\)-space and we have written the change in macroscopic inverse dielectric constants using the Lyddane-Sachs-Teller relation. In this way, for a given LO-TO phonon pair, we have a separate contribution from each phonon, since both \(\alpha_P\) and the dielectric constant factor depend on the phonon considered. We can thus estimate the effect for each phonon and add them, thereby generalizing Nery and Allen’s simple model to the case of multiple phonons. In SrTiO\(_3\), there are three optically active phonons.

FIG. 2. Crystal structures of SrTiO\(_3\) a) cubic, b) tetragonal \(I4/mcm\), and c) layered orthorhombic CaIrO\(_3\) view from the z-axis.
The results of this approach and the corresponding parameters are summarized in Table IV. We can see that the conduction band is predicted to shift less than the valence band as expected and the total gap correction is predicted to be 404 meV, which we really should round off to 0.4 eV. The shortest polaron length scale corresponds to holes for the largest phonon frequency and is 8 Bohr. This corresponds to a q-space region of about 1/6 of the Brillouin zone. Our estimate using the BM approach in Table III used a 4 × 4 × 4 mesh and gives a contribution to the zero-point motion or lattice polarization correction of −0.55 eV. This is already rather close to the polaron estimate. With a 6 × 6 × 6 mesh we obtain −0.25 eV. These bracket the polaron estimate of Table IV. We can thus conservatively conclude that the lattice polarization correction amounts to 0.3 ± 0.1 eV in good agreement between the Nery-Allen like estimate (0.4) and the BM approach. When we add this to the 0.8Σ result we obtain a gap of 3.24 eV for the indirect gap in excellent agreement with experiment. We note that if we apply the lattice polarization correction using the BM approach with a 4 × 4 × 4 mesh but then apply the 0.8Σ correction, the LPC shift is also reduced by 0.8, and becomes 0.4 eV. We can see that in this approach the correction is almost a constant shift and hence the indirect gap correction is the same as the direct gap correction. Because of the approximate nature of these estimates, we have not separately evaluated the polaron
approach to the VBM at Γ which would give the direct gap. In principle, the polaronic effect also should enhance the band mass by a factor \((1 + \alpha P/6)\) but it is not clear that the BM-method captures this more subtle effect. In fact, we find the bands to shift almost rigidly as can be seen in Fig. 4.

C. Other band structure features

Turning to other band features than the gap, summarized in Table V we see that Sr-4p states lie significantly closer to the VBM than the Ti 3p semicore states and hence play a more important role. We can see that the shifts of these states are also sensitive to the 0.8Σ and LPC corrections and amount to about 2 eV for Sr-4p and 4 eV for Ti 3p. As expected, the farther away from the VBM, the larger is the quasiparticle self-energy shift. In the conduction band we see that the higher lying \(\Gamma_{12}\) state has almost the same shift from LDA (about 1.4 eV) as the \(\Gamma_{25}'\) CBM. In the valence band the shifts are smaller and progressively larger as we go deeper in the VBM.

D. Tetragonal structure

The band structure for the tetragonal structure is shown in Fig. 3b. In the tetragonal material we see a similar large shift of the band gap by GW. To understand this band structure, we note that the tetragonal unit cell is rotated by 45° and has \(a_t = \sqrt{2}a_c\) as in-plane lattice constant. Thus the Brillouin zone (BZ) of the cubic structure is folded into a smaller BZ with the \(\Gamma - M\) of the tetragonal BZ corresponding to half the \(\Gamma - X\) of the cubic BZ. The high symmetry points correspond to \(M = (1/2, 1/2, 0)\) and \(X = (1/2, 0, 0)\) with respect to their respective reciprocal lattice vectors. Similarly the \(\Gamma - X\) of the tetragonal BZ is half the \(\Gamma - M\) of the cubic BZ. One can clearly see the folding in half of the bands with additional small gaps opening due to the breaking of the symmetry by the slight rotation of the octahedra. We can see that VBM which in the cubic case and in QSGW occurs between \(M - R\) (\(R = (1/2, 1/2, 1/2)\)), where the band dispersion is very flat, is folded on to the tetragonal
TABLE V. Various band differences: specific states (symmetry labeling as in Fig. 3a) in the conduction band relative to the CBM, and in the upper valence band relative to the VBM. The position of the semicore states Sr4p, Ti3p, are also with respect to the VBM at Γ.

|         | Γ25′-CBM | Γ12- CBM | Γ15- VBM | Γ25- VBM | Γ25- VBM | Sr4p states | Ti3p states |
|---------|----------|----------|----------|----------|----------|-------------|-------------|
| LDA     | 1.74     | 4.29     | -0.41    | -1.12    | -2.91    | -14.66      | -32.55      |
| G0W0    | 3.53     | 6.29     | -0.58    | -1.37    | -3.49    | -17.19      | -36.48      |
| QSGW    | 4.34     | 6.98     | -0.49    | -1.39    | -3.67    | -17.51      | -37.38      |
| 0.8Σ-QSGW | 3.54   | 6.17     | -0.49    | -1.32    | -3.64    | -16.9       | 36.74       |
| QSGW+LP | 3.78     | 6.4      | -0.49    | -1.37    | -3.62    | -17.51      | -37         |
| QSGW+LP+0.8Σ | 3.2 | 5.71     | -0.49    | -1.31    | -3.5     | -16.9       | -36.4       |

TABLE VI. Band gap of tetragonal SrTiO3 in eV at Γ − Γ.

| basis-sets | LDA | QSGW | QSGW+LP | QSGW+0.8Σ | Ref. |
|------------|-----|------|---------|-----------|------|
| with Sr4p,Ti3p | 1.76 | 4.1  | 3.88    | 3.27      | 4.01 |
| no semi-core | 1.72 | 3.7  | 3.47    | 3.04      |      |

a, b By Heifets and et al.\cite{13} using a hybrid DFT-Hartree-Fock (HF) approach

BZ Γ point and the gap becomes direct.

E. Hypothetical layered orthorhombic structure

Although, the CaIrO3 structure, proposed\cite{23} for SrTiO3 as a potential high-pressure structure, can be shown to be unstable,\cite{23} it is of interest to see how the GW gap correction changes with such a large change in structure. This structure has edge-sharing octahedra in layers separated by Sr, rather than corner sharing octahedra. In the LDA, the band gap becomes zero as can be seen in Fig. 3. The very different band dispersion in this case results from the direct Ti-d to Ti-d interactions between much closer Ti atoms in the layer. In the QSGW method the gap becomes 2.32 eV which is not too different from the gap correction 2.68 eV in cubic perovskite. The gap correction is found to be almost the same as in the cubic or tetrahedral structures. Similar screening reduction or 0.8Σ corrections and lattice polarization corrections should apply here but are not further pursued at this point.

VII. CONCLUSIONS

In this paper we reviewed the status of the QSGW method for a prototypical complex transition metal oxide like SrTiO3 in the perovskite structure. We found that all-electron QSGW results obtained by means of the FP-LMTO implementation give a significant overestimate of the gap compared to experiment in contrast to PAW or pseudopotential based GW approaches. This indicates a compensation of errors in the latter. We base this on the observation that for a large family of materials, the under-screening of W in the RPA amounts to about 20 % and can hence be accommodated by using the 0.8Σ approach. This evidence is based both on the comparison of dielectric constants in QSGW with experiment and on recent calculations\cite{11,12} which go beyond the RPA by including an exchange correlation kernel in the calculation of W or adding vertex corrections directly\cite{12} and it is found to apply to both tetrahedrally bonded semiconductors and various oxides and ionic compounds. The second important correction to the gap is the lattice polarization correction. This is part of the zero-point motion correction due to electron-phonon coupling and more specifically is its dominant contribution in strongly ionic materials arising from the long-range Fröhlich part of the electron-phonon coupling. Two independent estimates of this effect were made: one based on the polaron theory and one on the Botti-Marques approach of multiplying the macroscopic dielectric constant at q = 0 by a Lyddane-Sachs-Teller factor along with a suitable q-mesh sampling based itself on the polaron length scale which determines the strength of the effect. The two estimates are found to be in good agreement with each other. We find that both the electron-hole interaction effects which reduce Σ by about 20% and the lattice-polarization corrections are required to obtain good agreement with experimental gaps in cubic perovskite SrTiO3. As for the structural dependence of the QSGW corrections, we find that the gap correction in tetragonal STO is very close to that in cubic STO and the bands are essentially folded according to the rotation of the octahedra, which leads to a doubling of the cell and rotation of the BZ by 45°. This happens to fold the R point of the BZ onto the Γ-point an hence the indirect lowest gap becomes then direct. Due to the similarity in band states, we expect it to be pseudo-direct in the sense that no strongly optical transitions will correspond to this direct gap. Even for a very different hypothetical structure with edge-sharing octahedra, we find very similar gap corrections by QSGW, which shows that the gap corrections are rather insensitive to structure.
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