ABSTRACT: Urban air pollution is a critical health problem in cities all around the world. Therefore, spatially highly resolved real-time monitoring of airborne pollutants, in general, and of nitrogen dioxide, NO₂, in particular, is of utmost importance. However, highly accurate but fixed and bulky measurement stations or satellites are used for this purpose to date. This defines a need for miniaturized NO₂ sensor solutions with detection limits in the low parts per billion range to finally enable indicative air quality monitoring at low cost that facilitates detection of highly local emission peaks and enables the implementation of direct local actions like traffic control, to immediately reduce local emissions. To address this challenge, we present a nanoplasmonic NO₂ sensor based on arrays of Au nanoparticles coated with a thin layer of polycrystalline WO₃, which displays a spectral redshift in the localized surface plasmon resonance in response to NO₂. Sensor performance is characterized under (i) idealized laboratory conditions, (ii) conditions simulating humid urban air, and (iii) an outdoor field test in a miniaturized device benchmarked against a commercial NO₂ sensor approved according to European and American standards. The limit of detection of the plasmonic solution is below 10 ppb in all conditions. The observed plasmonic response is attributed to a combination of charge transfer between the WO₃ layer and the plasmonic Au nanoparticles, WO₃ layer volume expansion, and changes in WO₃ permittivity. The obtained results highlight the viability of nanoplasmonic gas sensors, in general, and their potential for practical application in indicative urban air monitoring, in particular.

KEYWORDS: nanoplasmonic sensor, WO₃, NO₂, air quality, parts per billion, urban air

Ensuring a healthy and livable urban environment is a priority all over the world due to rapidly progressing urbanization. According to the WHO, air pollution, in general, and nitrogen dioxide (NO₂), in particular, are among the largest health risk factors. As a consequence, the real-time monitoring of airborne pollutants, such as NO₂, is of utmost importance to reliably assess their impact, to enable crafting and accurate evaluation of new policies, and for decision makers to take fast action in response to local air pollution episodes, such as real-time traffic congestion control. To monitor air quality, to date, highly accurate but costly, stationary and bulky measurement stations are used, and chemiluminescence has been defined as the standard NO₂ measurement method in the corresponding European Standard (EN 14211: 2012). The data gathered by such monitoring stations provide high accuracy but offers only very low spatial resolution since these stations are very sparsely deployed at a few locations only due to their high cost. Hence, deeper insights into highly resolved spatial and temporal variability of pollutants remain impossible. Consequently, a technological breakthrough enabling—ideally—equally accurate but mobile and spatially highly resolved air quality monitoring devices are needed. To this end, one of the remaining key challenges is the required detection limit for NO₂ in the low parts-per-billion (ppb) range and in the presence of potentially interfering molecular species abundant in urban air, such as O₂, CO₂, CO, and H₂O. Therefore, significant research has been invested in developing NO₂-sensing platforms comprising different materials and utilizing different readout principles, as summarized in recent reviews. Among the NO₂-sensitive materials, metal oxides, in general, and tungsten trioxide (WO₃), in particular, have been identified as highly NO₂-selective and have therefore been explored in a plethora of designs, ranging from thin films to colloidal nanoparticles. Among a large number of sensor readout principles, resistive metal-oxide-semiconductor (MOS-type) sensors and electrochemical sensors are to date considered the best compromise in terms of technology maturity, sensitivity, cost, and device miniaturization potential. However, the...
performance of MOS-type sensors is limited by their long response time and signal drift, whereas electrochemical sensors are limited by cross-sensitivity and susceptibility toward changes in the humidity level and temperature. At the same time, nanoplasmonic gas sensors based on localized surface plasmon resonance (LSPR) have recently emerged as a competitive technology platform with high sensitivity, fast response, and significant miniaturization potential, in principle, down to the level of the individual nanoparticle. In the context of NO2 sensing, a proof-of-principle plasmonic detection combined with an NO2-selective material, such as a metal oxide or a molecular compound, has been demonstrated. However, no reports about the application of plasmonic NO2 sensors in real urban air exist, and their limit of detection (LoD) is generally widely unexplored.

Here we report a nanoplasmonic NO2 sensor platform based on arrays of Au nanoparticles coated with a thin layer of highly polycrystalline WO3 for which we assess in detail its response to NO2 under (i) idealized laboratory conditions, (ii) conditions simulating humid urban air and (iii) in a realistic field test in the city of Göteborg, Sweden, benchmarked with a stationary chemiluminescence-based nitrogen oxide analyzer (Serinus 40, Acoem). As the key results, we find an extrapolated sensor LoD of about 3 ppb in all conditions, including the field test. This performance exceeds or is on par with the most sensitive NO2 sensors reported in the literature. Furthermore, together with the highly promising field test results, our findings highlight the potential of nanoplasmonic air quality sensors for large-scale deployment in urban environments for the purpose of so-called indicative monitoring of urban air. Such indicative monitoring serves the purpose of identifying the periods and spatial distribution of elevated NO2 concentrations with high spatial resolution and, therefore, to be seen as a complement to, rather than a replacement of, the highly accurate measurement stations used to date.

RESULTS AND DISCUSSION

Sensor Nanofabrication and Characterization. The sensor surfaces were prepared by nanofabricating a quasirandom array of Au nanodisks 120 nm in diameter and 20 nm in thickness onto a transparent Borofloat glass substrate and encapsulated by a 40 nm thick WO3 film support (Boro float, Schott Scandinavia AB) using Hole-mask Colloidal Lithography (details in Methods). To functionalize it for NO2 detection with high specifity, we deposited a 40 nm thick WO3 film onto the nanostructured surface by RF magnetron sputtering, followed by two-step annealing at 400 °C for 12 h in 4% H2 in Ar, and subsequently at 400 °C for 12 h in air. This resulted in Au nanoparticles completely encapsulated in a highly polycrystalline layer (Figure 1a,b), for which X-ray photoelectron spectroscopy (XPS) analysis reveals that the W 4f7/2 and W 4f5/2 doublet peaks, whose maxima are positioned at 36.7 ± 0.1, 38.8 ± 0.1 eV, respectively, which is in good agreement with a WO3 surface oxidation state. (d) Optical extinction spectra of a nanoplasmonic Au–WO3 sensor before (blue) and after (red) exposure to 1 part per million (ppm) NO2 in dry synthetic air. The interaction with NO2 induces a spectral redshift, Δλpeak, of the LSPR peak.

Figure 1. (a) Schematic top view and cross-section through the Au–WO3 nanoplasmonic sensor depicting the quasirandom array of Au nanodisks fabricated onto a transparent Borofloat glass substrate and encapsulated by a 40 nm thick WO3 film. (b) SEM image of a sensor surface revealing the WO3-coated Au nanodisks and the highly polycrystalline WO3 coating. Scale bar equals 100 nm. (c) High-resolution XPS spectrum of the annealed sensor surface in the energy region of the W 4f7/2 and W 4f5/2 doublet peaks, whose maxima are positioned at 36.7 ± 0.1, 38.8 ± 0.1 eV, respectively, which is in good agreement with a WO3 surface oxidation state. (d) Optical extinction spectra of a nanoplasmonic Au–WO3 sensor before (blue) and after (red) exposure to 1 part per million (ppm) NO2 in dry synthetic air. The interaction with NO2 induces a spectral redshift, Δλpeak, of the LSPR peak.
NO$_2$ is enhanced in the presence of the oxygen vacancies since the common conclusion is that the interaction between WO$_3$ and other oxides like ZnO, SnO$_2$, and In$_2$O$_3$ conductivity of the WO$_3$-sensing layer in the presence of NO$_2$ are of the MOS-type, in which measured changes in the range from 100 to 300 °C, according to the following principles. These principles all have in common that they exploit the fact that oxygen molecules strongly interact with metal oxide surfaces, in general, and with WO$_3$, in particular, according to the following scheme

\[
\text{O}_2(\text{gas}) \rightarrow \text{O}_2(\text{ads}) 
\]

\[
\text{O}_2(\text{ads}) + e^- \rightarrow \text{O}^- (\text{ads}) 
\]

\[
\text{O}^- (\text{ads}) + e^- \rightarrow 2\text{O}_2^- (\text{ads}) 
\]

Here, depending on the operating temperature, different oxygen species are predominantly adsorbed on a WO$_3$ surface, that is, for temperatures below 100 °C, it is mostly O$_2^-$ that captures electrons from the WO$_3$ conduction band, and in the range from 100 to 300 °C, oxygen is mainly adsorbed in the form of O$^-$. (Figure 2a,b). Introducing also NO$_2$ to the system leads to the coadsorption of O$_2$ and NO$_2$ ions. However, owing to the five times higher electron affinity of NO$_2$ compared to O$_2$, NO$_2$ chemisorbs in the forms of NO$_2^-$ (nitrite ion) or NO$_3^-$ (nitrate ion) by capturing electrons either from WO$_3$ or from preadsorbed oxygen species, according to the following reactions

\[
\text{NO}_2 + e^- \leftrightarrow \text{NO}_2^- 
\]

\[
\text{NO}_2 + \text{O}^- \rightarrow \text{NO}_3^- 
\]

Since thereby an electron transfer from the surface to the analyte molecules takes place, the electrical conductivity of the active metal-oxide-sensing layer is altered, enabled by the existence of native vacancies and defects in its structure. The specific role of these defects in WO$_3$-based NO$_2$ detection has been investigated in detail in various studies. The common conclusion is that the interaction between WO$_3$ and NO$_2$ is enhanced in the presence of the oxygen vacancies since they function as active adsorption sites for NO$_2$. Consequently, the majority of reported WO$_3$-based NO$_2$ sensors are of the MOS-type, in which measured changes in the conductivity of the WO$_3$-sensing layer in the presence of NO$_2$ constitute the signal transduction principle. Accordingly, also other oxides like ZnO, SnO$_2$, and In$_2$O$_3$ have been used in MOS-type NO$_2$ sensors by exploiting a similar detection principle.

In this study, however, we utilize a different sensing principle, which on the one hand relies on the strong interaction of the Au nanodisk array on the sensor surface with incident visible-NIR light via LSPR, and on the other hand, the sensitivity of the LSPR to changes occurring both to the plasmonic nanoparticles themselves and to their intimate surroundings, which subsequently is reflected in a finite $\Delta \lambda_{\text{peak}}$ (cf. Figure 1d). To specifically rationalize the origin of the observed $\Delta \lambda_{\text{peak}}$ signal generated by NO$_2$ for the sensor surface at hand, we recall that the LSPR frequency of a Au nanoparticle, $\Omega$, in its simplest form, is a function of the free electron density in the metal and the refractive index of the surrounding matrix as

\[
\Omega = \sqrt{\frac{N\varepsilon^2}{(1 + 2m_e)m_0\varepsilon}} 
\]

where $N$ is the conduction electron density, $\varepsilon$ is the elementary charge, $m_0$ is the dielectric function of the matrix, $m_1$ is the electron mass, and $\varepsilon_0$ is the permittivity of free space. Translated to the case at hand, the chemisorption of NO$_2$ onto the WO$_3$ surface leads to a conductivity change of the WO$_3$ layer due to electron depletion by the formed NO$_2$ species on its surface, as discussed above. Consequently, owing to a subsequent charge equilibration between the WO$_3$ layer and the Au nanoparticles, the free electron density of these particles is slightly reduced and leads to the observed spectral redshift of the LSPR peak, as also proposed in the literature for other Au—metal oxide nanocomposite plasmonic gas sensors.

Next, it is also interesting to briefly consider the likely impact of NO$_2$ concentration in the analyte medium on this process. For low NO$_2$ concentrations in the ppb range, the equilibrium coverage of NO$_2^-$ is low and likely limited to the surface, rendering charge transfer from the Au nanodisks to surface-bound chemisorbed NO$_2^-$ via WO$_3$, the main sensing mechanism (Figure 2c). However, when the NO$_2$ concentration in the analyte medium increases to the parts per million (ppm) range, the equilibrium NO$_2^-$ and NO$_3^-$ coverage on the sensor surface increases significantly, and the formation of

![Figure 2. Schematic depiction of the proposed NO$_2$-detection mechanism of Au—WO$_3$ nanoplasmonic sensors.](https://doi.org/10.1021/acssensors.1c02463)
Nitrogen dioxide ($NO_2$) is favored as observed on various metal oxides in experimental studies and corroborated by theoretical calculations. Since adsorbed $NO_2$ species are also known to have a higher stability than adsorbed $NO_3$, it becomes increasingly likely that a subsurface transformation of $WO_3$ into $W(NO_2)_x$ also takes place at high $NO_2$ concentrations in the analyte medium (Figure 2c). Since this process not only leads to a charge transfer but also induces a volume expansion and a sizable change in permittivity of the oxide matrix around the Au nanoparticles (both of unknown magnitude since no corresponding studies determining their magnitude exist to the best of our knowledge), the observed plasmonic response at higher $NO_2$ concentrations is likely a cumulative effect of three factors, that is, (i) charge transfer, (ii) matrix volume expansion, and (iii) matrix permittivity change (Figure 2c).

In addition, we note that it is likely that the sputtered $WO_3$ layer exhibits a certain degree of porosity. In principle, this means that such pores may enable $NO_3$ diffusion to the Au/ $WO_3$ interface and thus direct interaction between Au and $NO_3$ that may contribute to or even provide a complementary sensing mechanism. However, as our control experiments on uncoated Au nanoparticles reveal, even at high $NO_2$ concentrations in the 5–10 ppm range, no significant $\Delta \lambda_{\text{peak}}$ response is recorded (Figure S1), which corroborates the sensing mechanism discussed above.

**$NO_2$ Detection in Dry Synthetic Air.** To test the sensing performance toward $NO_2$ in dry laboratory conditions, we first conditioned an as-fabricated and thermally annealed sensor by exposing it for 4 h to synthetic air at 250 °C. After this conditioning stage, we conducted $NO_2$-sensing measurements from the 1 ppm down to 15 ppb $NO_2$ concentration range (the lowest concentration attainable with our setup) by exposing the sensor to different $NO_2$ pulses with different concentrations in synthetic air at 250 °C (Figure 3a). Each concentration step was repeated 3–4 times (Figure S2).

Evidently, the sensor exhibits a consistent, reversible, and reproducible response that distinctly depends on $NO_2$ concentration. Furthermore, a typical noise level of $\sigma = 0.006$ nm can be extracted from the sensor response (Figure 3b). To determine the concentration dependence of this response and thereby generate a calibration curve, we extracted $\Delta \lambda_{\text{peak}}$ for all measured $NO_2$ pulses and plot them as a function of $NO_2$ concentration (Figure 3c). This analysis reveals a distinct concentration dependence of $\Delta \lambda_{\text{peak}}$ and an extrapolated LoD of 3 ppb at these idealized dry conditions in synthetic air. It is also worth noting that the error bars at higher $NO_2$ concentrations are larger than at lower concentrations. This is likely the consequence of our measurement sequence implemented from high to low $NO_2$ concentration (Figure S2) since the sensor is “fresh” at the first high concentration exposures and, therefore, initially undergoes a certain degree of structural conditioning during the first exposures to $NO_2$ before reaching a new morphological equilibrium state.

**Temperature Dependence of Sensor Response in Dry Synthetic Air.** The operating temperature has been reported to have a significant impact on the $NO_2$-sensing performance of $WO_3$. Hence, it is important to characterize our system in this respect. To do so, we investigated the sensors in dry synthetic air in the temperature range from 50 to 250 °C, with 50 °C increments, using both the highest and lowest $NO_2$ concentrations of our measurement range, that is, 1 ppm and 15 ppb. Focusing first on the high concentration 1 ppm pulses,

\[ \Delta \lambda_{\text{peak}} \text{ increases significantly with temperature up to } 200 \text{ °C.} \]

Then, we don’t observe a further $\Delta \lambda_{\text{peak}}$ increase when ramping up the operating temperature to 250 °C (Figure 4a). Interestingly, a different trend is revealed for the 15 ppb case, for which we record no response at 50 °C and a maximum amplitude at 150 °C before decreasing again at even higher temperatures (Figure 4b).

To rationalize the identified significantly different temperature dependencies of the sensor at 15 ppb and 1 ppm (Figure 4b), we propose a concept of “equilibrium states” of the analyte medium. In general, the $NO_2$ concentration step was repeated 3–4 times (Figure S2).
A Speciﬁc experiment that closely resembles real ambient conditions. To this end, while a detailed assessment of the role of these different molecular species in the sensing process is beyond the scope of our study, an earlier study has revealed complex surface chemistry as a consequence of the fact that CO, H₂O, and NO₂ are oxidizing, whereas CO is a reducing gas. This, for example, means that they either may compete for or assist with the adsorption of NO₂ on the surface. After this point, the former effect is expected to be most prominent in the low NO₂ concentration regime, where the sensor response is expected to be solely dictated by NO₂ coverage on the surface, and thus explains why we observe a signal amplitude maximum at 150 °C (Figure 4b,c). At higher NO₂ concentrations in the ppm range, on the other hand, the temperature dependence of the NO₂ surface coverage is expected to be signiﬁcantly less pronounced as the surface is expected to be completely covered in the considered temperature range. Therefore, in this regime, reaction kinetics for the formation of W(NO) x become more relevant and the dominating factor that dictates the sensor response amplitude, thereby explaining the observed continuous Δλ_peak increase for the increasing temperature at 1 ppm NO₂ as well as the generally accelerated response (Figure 4a,c). As the main conclusion, we thus identify a sensor operation temperature of 150 °C as the best compromise for a wide dynamic range and use it from here forward.

**NO₂ Detection in Simulated Humid Urban Air.** To further benchmark our nanoplasmonic Au–WO₃ sensor platform for air quality monitoring in urban air, we designed an experiment that closely resembles real ambient conditions. Specifically, we operated the system in synthetic air mixed with 1 ppm CO and 400 ppm CO₂, humidified to 50% relative humidity (RH) at 30 °C, to emulate urban air at ambient conditions, where the CO and CO₂ concentrations mimic the natural abundance of these species. Like in the previous experiments, we then exposed the sensor to NO₂ pulses at concentrations ranging from 1 ppm down to 15 ppb (Figure S3), with the sensor heated to 150 °C that we identiﬁed above as the best compromise in terms of sensitivity toward both high and low NO₂ concentrations (Figure 5a). As the main result, we observe a distinct, reversible, and NO₂ concentration-dependent Δλ_peak response down to 15 ppb, which again is the lowest concentration we can produce in our setup. This is a remarkable performance since it is achieved despite potential cross-sensitivity to the background species in the gas mixture.

To this end, a detailed assessment of the role of these different molecular species in the sensing process is beyond the scope of our study, an earlier study has revealed complex surface chemistry as a consequence of the fact that CO, H₂O, and NO₂ are oxidizing, whereas CO is a reducing gas. This, for example, means that they either may compete for or assist with the adsorption of NO₂ on the surface. After this point, the former effect is expected to be most prominent in the low NO₂ concentration regime, where the sensor response is expected to be solely dictated by NO₂ coverage on the surface, and thus explains why we observe a signal amplitude maximum at 150 °C (Figure 4b,c). At higher NO₂ concentrations in the ppm range, on the other hand, the temperature dependence of the NO₂ surface coverage is expected to be significantly less pronounced as the surface is expected to be completely covered in the considered temperature range. Therefore, in this regime, reaction kinetics for the formation of W(NO) x become more relevant and the dominating factor that dictates the sensor response amplitude, thereby explaining the observed continuous Δλ_peak increase for the increasing temperature at 1 ppm NO₂ as well as the generally accelerated response (Figure 4a,c).
sensor is exposed sequentially to them 9,33), thereby truly where all molecular species are mixed (and not where the key distinctive feature and a step beyond this state of the type NO2 sensors reported in the literature.9,33 However, as the last step of our Au quality sensor device (Insplorion AB, Gothenburg, Sweden) to benchmarking, we integrated it with a miniature urban air environment in Gothenburg, Sweden, over the span of 5 days by mounting the device close to a road with high traffic activity in the city (Figure 6a). As the main result, we obtained reliable real-time NO2 concentration measurements by the plasmonic NO2 sensor in a concentration range of 25–100 ppb, with a general rise of the ambient NO2 levels during daytime and with distinct peaks due to increased traffic activity (Figure 6d—all data have been averaged to 15 min increments). Remarkably, the measured general trends and absolute concentration values are in quite good agreement with the reference measurements executed simultaneously using the Serinus 40 reference station.

At the same time, we observe some discrepancies in the quantification of NO2 concentrations for some measurement periods. To put these into perspective, we first note that the Serinus 40 is a certified reference instrument that detects NO2 by chemiluminescence with high accuracy, whereas our sensor device has been developed with the intention to be used for indicative monitoring. In this sector, to date, no standards exist, and lower accuracy can be tolerated as a trade-off for the possibility to deploy miniaturized and cost-effective sensors with high spatial density across, for example, a city.

Nevertheless, despite this difference in scope of the two systems, it is important to discuss the potential reasons for the observed discrepancies. As the contributing first reason, we identify the different gas intake characteristics of the two systems. In the plasmonic system, the sensor surface is separated from the ambient air by a polytetrafluoroethylene membrane with 100 nm pore size, which means gas transport to the sensor surface is entirely reliant on diffusion, with the membrane being the bottleneck. The Serinus 40, in contrast, uses pneumatic ports for air sampling, which very likely creates different mass transport characteristics in the two systems, and, therefore, affects response and recovery times on the test its NO2-detection performance in real urban air in a proper field test. To generate the sensor readout, the device measures the relative change in transmitted light intensity by the sensor chip over a range of wavelengths in the red/NIR spectral region. The specific wavelength range is chosen to coincide with the left flank of the LSPR peak of the sensors to maximize the transmittance change upon a shift of the peak9,10 induced by a change in NO2 concentration. To measure this transmittance change, standard light-emitting diodes and surface-mounted photodetectors are used in the device, and a microcontroller maintains the working temperature of the sensor chip constant at above 100 °C. The fractional increase in light transmitted through the sensor chip, caused by a redshift of the LSPR peak, is used as the signal readout.

To calibrate the device prior to the field test measurements, we exposed it to multiple pulses and steps of NO2 in dry synthetic air in the concentration range of 25–100 ppb in the laboratory (Figure 6a). The obtained response plotted as a calibration curve is shown in Figure 6b. It indicates an extrapolated LoD of 2.0 ppb, which is on par with the LoD’s identified above for the sensor chips alone and using Δλpeak as the readout (cf. Figures 3c and 5c). Based on this calibration curve, a transfer function relating the change in relative transmittance measured by the device and NO2 concentration was determined. The microcontroller in the device was then configured to automatically perform the transfer function during the field measurements to determine the NO2 concentration in real time.

The field test itself was conducted by sampling air from an urban environment in Göteborg, Sweden, over the span of 5 days by mounting the device close to a road with high traffic activity in the city (Figure 6c). As the main result, we obtained reliable real-time NO2 concentration measurements by the plasmonic NO2 sensor in a concentration range of 2–25 ppb, with a general rise of the ambient NO2 levels during daytime and with distinct peaks due to increased traffic activity (Figure 6d—all data have been averaged to 15 min increments). Remarkably, the measured general trends and absolute concentration values are in quite good agreement with the reference measurements executed simultaneously using the Serinus 40 reference station.

To put this result into perspective, we first note that an LoD of 3 ppb is on par with the best thin-film WO3-based MOS-type NO2 sensors reported in the literature.9,33 However, as the key distinctive feature and a step beyond this state of the art, our sensors exhibit this low ppb LoD in an environment where all molecular species are mixed (and not where the sensor is exposed sequentially to them9,33), thereby truly emulating a real urban air environment.

Field Testing a Nanoplasmonic Au–WO3 NO2 Sensor.

As the last step of our Au–WO3 nanoplasmonic sensor chip benchmarking, we integrated it with a miniature urban air quality sensor device (Insplorion AB, Göteborg, Sweden) to

Figure 5. (a) Time-resolved Δλpeak response of a nanoplasmonic Au–WO3 sensor upon exposure to different NO2 concentration pulses in synthetic air mixed with 1 ppm CO, 400 ppm CO2, and 50% RH set at 30 °C. The sensor operating temperature was 150 °C. The shaded area denotes the NO2 pulse duration. (b) Zoom-in on the time-resolved Δλpeak response of the sensor to 15 ppb NO2 from (a). Inset: noise level determination revealing a standard deviation (σ) of 0.005 nm, as denoted by the red band. (c) Δλpeak of the sensor plotted as a function of NO2 concentration. The error bars denote the standard deviation from three pulses at each NO2 concentration. The solid line depicts a fit to the experimental data using the Redlich–Peterson semiempirical adsorption model.64 The inset shows the same plot for the low end of the NO2 concentration range. The green- and red-dashed lines signify the threefold noise level (3σ = 0.015 nm) and the extrapolated limit of detection (LoD = 3.1 ppb), respectively.

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As the last step of our Au–WO3 nanoplasmonic sensor chip benchmarking, we integrated it with a miniature urban air quality sensor device (Insplorion AB, Göteborg, Sweden) to
shorter time scales. These effects are, however, not severe enough to explain the major observed discrepancies between the two systems.

A second potentially important factor to consider is varying humidity during the field test due to weather variations in the course of the 5 day period. Here, in the Serinus 40, water is removed from the sampled air by Naion tubing inside its dryer compartment, and the instrument, thus, always samples dry air, whereas in the plasmonic device, the ambient air is sampled as is. Hence, even though relative humidity changes occurring at ambient conditions due to weather variations are reasonably small when translated to the plasmonic sensor’s high operating temperature, they are likely still relevant. This hypothesis is corroborated by our laboratory measurements in humid synthetic air, which revealed faster response with larger amplitude per unit NO2 in humid (cf. Figure 5a) compared to dry (cf. Figure 3a) conditions. This, thus, suggests that (a part of) the discrepancy between the two sensor systems used in the field test may be the consequence of humidity variations.

As a final aspect, we note that in urban air, not only NO2 but also NO is present, however, usually at even lower concentrations. Therefore, it is relevant to briefly address the potential cross-sensitivity of our plasmonic sensor toward NO. Here, we can resort to that Serinus 40 also measured the NO concentration during the field test, yielding an average of a few 10−ppb NO concentrations. (Figure S4). Furthermore, a control experiment, where we exposed the plasmonic sensor to 2 and 3 ppm NO, revealed an opposite response, that is, a spectral blueshift of the LSPR peak (compared to a redshift for NO2) at essentially one order of magnitude smaller amplitude compared to the corresponding response to NO2 (Figure S5). This finding is in-line with similar studies performed with resistive metal oxide sensors and implies that variations in the NO concentration are likely negligible in the plasmonic sensor response in the NO concentration range identified for the field test and thus for air quality monitoring in general.

**CONCLUSIONS**

In conclusion, we have presented a Au−WO3 nanoplasmonic NO2 sensor with a sub-10 ppb limit of detection both in laboratory conditions and in a 5 day field test next to a highly trafficked road in Göteborg, Sweden, using a miniaturized autonomous sensor device, which we also benchmarked with a chemiluminescence-based Serinus 40 reference system certified both according to European (EN14211) and US EPA (RFNA-0809-186) standards. The found performance of the Au−WO3 nanoplasmonic NO2 sensor, which is enabled by a nanofabricated sensor chip surface comprising a quasirandom array of Au nanodisks coated with a 40 nm thick polycrystalline WO3 film operated above 100 °C, is on par with or exceeds the performance of existing solutions using alternative readout principles in terms of the limit of detection. The identified discrepancies between the plasmonic sensor and the reference system during the field test are identified as likely consequences of humidity variations handled differently by the two systems and highlight the importance of further investigations of humidity-related effects. Taken all together, these results prove the viability of nanoplasmonic gas sensors, in general, and their potential for practical application in indicative urban air monitoring, in particular, where low cost and large-scale deployment capability are the key enabling factors.
**METHODS**

**Sensor Nanofabrication.** Au nanodisk arrays were fabricated using the Hole-Mask Colloidal Lithography technique, which is described in detail elsewhere, onto 9.5 × 9.5 mm² glass substrates (BoroFloat, Schott S Chandina) and silicon wafer substrates (for SEM imaging and XPS measurements). In brief, the hole-mask nanofabrication steps were as follows:

1. Substrates were cleaned in an ultrasonic bath consecutively with acetone, isopropanol, and deionized water. Each step was applied for 3 min.
2. A PMMA (MicroChem, 950 000 molecular weight, 2 wt % in anisole) layer was spin-coated at a spin rate of 2000 rpm for 45 s. Subsequently, the substrate was placed on a hot plate at 170 °C for 5 min for soft-baking.
3. To reduce the hydrophobicity of the surface before drop-coating a suspension of positively charged poly-(diallyldimethylammonium chloride) (PDDA) solution, the substrates were exposed to oxygen plasma (Plasma-Therm Batchtop RIE 95 m, 50 W, 250 mTorr, 10 sccm) for 5 s.
4. A PDDA solution (Sigma-Aldrich, Mn = 200,000–350,000, 0.2 wt % in Milli-Q water) was drop-cast onto the PMMA layer and incubated for 45 s, followed by rinsing in deionized water and blow-drying with nitrogen gas.
5. A suspension of negatively charged polystyrene (PS) spheres (Interfacial Dynamics Corporation, 120 nm in diameter, 0.2 wt % in Milli-Q water) was drop-cast and incubated for 3 min, followed by rinsing in deionized water and blow-drying with nitrogen gas.
6. A 15 nm thick chromium film was deposited by e-beam physical vapor deposition (Lesker PVD 225, base pressure of 5 × 10⁻⁷ Torr, deposition rate of 1 Å/s).
7. The PS spheres were removed from the surface by tape-stripping (SWT-10 tape, Nitto Scandinavia AB) to reveal holes in the Cr film at the positions of spheres.
8. To complete the hole-mask pattern, the surface was exposed to oxygen plasma (Plasma-Therm Batchtop RIE 95 m, 50 W, 250 mTorr, 10 sccm) for 3 min to etch the PMMA layer through the holes in the Cr film.
9. A 20 nm thick gold film was deposited with the same technique and parameters used in step (6) to grow the Au nanodisks through the hole-mask.
10. The samples were soaked in acetone to dissolve the remaining PMMA layer, rinsed in isopropanol, and blow-dried in nitrogen gas. This final step left the surface covered with gold nanodisks.
11. A 40 nm thick WO₃ thin film was RF-magneton-sputtered onto the Au nanodisks using a power of 150 W and 1:1 Ar:O₂ (30 sccm) at 25 mTorr.
12. Two steps of annealing were applied as post-processing. The samples were annealed first at 400 °C for 12 h under the flow of 4% H₂ in Ar in a tube furnace, followed by 400 °C for 12 h in air.

**Material Characterization.** A Zeiss Supra 55 VP SEM was used for imaging sensor surfaces at an electron beam acceleration voltage of 10 kV using a secondary electron detector. For further material characterization, XPS measurements were executed in a PerkinElmer PHI 5000C ESCA system with an energy step width of 0.125 eV and a pass energy of 58.70 eV. The correction of peaks was done with respect to the carbon 1s peak using the Multipak 6.0 software.

**NO₂ Sensing Measurements.** The measurements were conducted in a quartz tube plug-flow reactor equipped with an optics unit for transmittance measurements (Insplover X1, Insplovision AB). The resistive heating coils around the tube and Eurotherm temperature controller enable measurements at up to 600 °C. The standard deviation of the sensor temperature reading is ~0.1 °C. The reactor was configured with several mass flow controllers (Bronkhorst ΔP) to regulate the gas compositions and with a humidifier (Bronkhorst-controlled evaporator and mixer) to mimic humid air. Synthetic air (Strandmøll AB, 20.9% O₂, 79.1% N₂) was used as the carrier gas, and all the gases involved in the measurements (NO₂, CO, CO₂—Strandmøll AB) were supplied from cylinders diluted in synthetic air. The total gas flow rate used in the experiments was 340 mL/min.

The sensor chip mounted in the reactor was illuminated by a tungsten halogen lamp (Avalight-Hal, Avantes) through an optical fiber with a collimating lens, and the transmitted light was collected using a fixed grating spectrophotometer (AvSpec-ULS2048CL-EVO, Avantes). A 20th degree polynomial fit is applied to the raw measured extinction spectra around the LSPR peak. The λₚₑ𝐚 kull is determined by finding the wavelength where the first derivative of the fitted polynomial is equal to zero. The shift in the λₚₑ𝐚 kull was used as the sensing descriptor in this study.

**Plasmonic NO₂ Sensor Device Field Test Measurements.** The sensor device was calibrated in laboratory settings prior to the field test measurements. The device was placed in a protective casing and mounted close to a road with high traffic activity in Göteborg, Sweden. The field test measurement was conducted over the span of 5 days. In order to compare the performance of the sensor device, the Senirus 40 nitrogen oxide analyzer was used to monitor the air in the vicinity of the device. NO₂ concentrations measured by the device and the analyzer were averaged to 15 min increments.

The Senirus 40 reference instrument uses the gas-phase chemiluminescence technique to detect NO and NO₂. The sample gas passes via two different paths—NO path and NO₃ path. The NO path has a longer residence time due to a delay loop and an NO₂ to NO converter. Any NO species passing through this path remains unaffected, whereas the NO₃ species are converted into NO. Hence, the total amount of NO reaching the reaction cell is the combination of original NO present in the sample and converted NO₂.

At the end of each path, the sample gas arrives at the reaction cell and reacts with ozone to form activated NO₂ species (chemiluminescence reaction for NO).

\[
\text{NO} + \text{O}_3 \rightarrow \text{NO}_2 + \text{O}_2 + \text{light} \tag{7}
\]

The luminescence of the activated NO₂ species is detected by a photomultiplier tube. The NO concentration is evaluated from the intensity of the chemiluminescence. The NO₂ concentration is calculated by subtracting the NO concentration obtained in the NO path from the NO₃ path.

The instrument holds both US EPA (RFNA-0809-186) and EN (EN14211) approval certificates.

**ASSOCIATED CONTENT**

**Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acssensors.1c02463.

Raw sensor data showing time-resolved Δλₚₑᵃ kull responses of Au nanodisks upon NO₂ exposure and the Au—WO₃ sensor upon NO₂ exposure in dry synthetic air and humid air, concentrations of NO₂ and NO measured using Senirus 40 during the field test, and response of the Au—WO₃ sensor upon NO exposure (PDF)

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Author Contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Notes

The authors declare the following competing financial interest(s): The corresponding author is scientific advisor at the company Insplorion AB who actively participated in this study.

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