Static and Dynamic Control of Fingerprint Landscapes of Liquid Crystal Network Coatings

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Supporting Information

ABSTRACT: Surface topography variations of liquid crystal networks in their functional coatings provide unique properties in these systems. Chiral-nematic polymer coatings self-organize in a fingerprint texture with the molecular helices parallel to the substrate with alternating domains of molecular units with parallel and perpendicular director orientation as controlled by the concentration of a reactive chiral additive. Driven by surface-tension differences and altered by anisotropic polymerization shrinkage, the coating may form hills and valleys hundreds of nanometers in size with different molecular alignment. The director orientation in the corrugations could be controlled by monomer diffusion during polymerization. Polymerization in the presence of a dichroic dye gives topographic elevations in which the molecules are oriented along the normal. Polymerization by means of a dichroic photoinitiator gives topographic elevations in which the molecules align parallel to the surface. By balancing the monomer diffusion and anisotropic polymerization shrinkage, relatively flat surfaces are also achieved. The different surfaces exhibit distinct topographical deformations when subjected to external stimuli, such as an AC electric field. This method can be universally extended to LC polymers with other alignment configurations.

KEYWORDS: dichroic molecules, surface topography, liquid crystal networks, fingerprint, electric field

1. INTRODUCTION

Functional surfaces that can dynamically change from one state to another have attracted intensive attention recently because of their applications in various functions, for instance, self-cleaning,1-4 haptics,5 directional movement, and transportation of objects.5-8 A large part of surface functionality is influenced by the topography of surface coatings and impressive functions are achieved with the aid of the surface topographical characteristics. As a next step, dynamic surface topographies actuated by external stimuli are anticipated to lead to new functions,9 such as switchable adhesion,9-10 tunable optical properties,11-14 and microfluidic manipulation.15,16 Convenient stimuli studied to actuate the surfaces can be either light,11-14 heat15-18 or electrical input.22,23 In particular, as anisotropic materials, liquid crystal networks (LCNs) and liquid crystal elastomers (LCEs) respond to the stimuli with order parameter reduction and affording mechanical deformation, contributing to their capability in performing various tasks.24-31 An important class of responsive LCNs are chiral-nematic polymer networks in which the average molecular orientation, defined by the director, described a helix. Especially when the helix axis is chosen to be parallel to the surface, interesting and unique fingerprint textures are formed that are capable of performing a number of tasks such as controlling friction, grip, and removal of dust and particles.4,8 However, in many cases, because of the Marangoni effect32 explaining surface-energy-related material transporta- tion and anisotropic shrinkage during polymerization,33 the fingerprint LC polymer coatings are decorated with intrinsic nonflat surface topographies, which are depending on the application sometimes desired and in other cases undesired. It

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is therefore important, despite many studies about the tuning of the orientation, handedness and pitch of the chiral nematic LC systems,34-39 to study the mechanism of the formation of the fingerprint topographies and to control them during the formation or by an active actuation after the formation of the LCN coating.

In this article, we first monitored the fingerprint topographical structure prior to and during polymerization of the chiral LCN using the digital holographic microscope (DHM). After understanding the mechanism of their formation, we demonstrate the control of the coating surface topography during its preparation by photopolymerization by means of dichroic absorption phenomena exploiting the periodic alternation of the molecular orientation in the fingerprint texture. Finally, we demonstrate switching by either an enhancement of the corrugations or an inversion of the structure by the application of an in-plane electrical field.

2. EXPERIMENTAL SECTION

2.1. Materials. Dichroic initiator 1-4'-heptyl-[1,1':4',1'-terphenyl]-4-yl)-2-methyl-2-morpholinopropan-1-one) was obtained from Philips Research Laboratories. Reactive liquid crystal monomers 2 (2-cyano-1,4-phenylene bis(4′-(6-(acryloyloxy)hexyl)oxy)benzoate), 3 (4-methoxyphenyl 4′-(6-(acryloyloxy)hexyl)oxy)benzoate), and 4 (6-(4′-cyano-1,1'-biphenyl)-4-yl)hexyl acrylate) were obtained from Merck GmbH. Chiral dopant 5 ((3R,3aS,6S,6aS)-hexahydrofuro[3,2-b]furan-3,6-diyliibis(4(4-(4(acryloyloxy)butoxy)carbonyloxy)benzoyloxy)benzoate) was purchased from BASF. Dichroic dye 6 (1-(4-butylphenyl)-2-(4-(4-propoxyphenyl)diazenyl)phenyl)diazen) was obtained from Hayashibara Biochemical Laboratories, INC., Japan. Initiator 7 ((1-hydroxy cyclohexyl)-(phenyl)methanone, IRGACURE 184) was purchased from Ciba. LC monomers were mixed in the weight ratio of 2:3: 4:10:50.40

2.2. Fabrication. Glass substrates were cleaned by sonication in acetone and isopropanol for 10 min, respectively, and dried under N2. The polyimide SE 7511L (Nissan Sunever) was spin-coated onto the substrate at a speed of 5000 rpm and then baked at 100 °C for 10 min and 200 °C for 90 min. The LC monomers mixture was dissolved in THF and spin-coated on the substrate. Sometimes, a unidirectional low-pressure buffing on the polyimide layer prior to processing LC monomer mixture was applied to create more regular line structures. Subsequently, the coating was photopolymerized in a N2 atmosphere by UV illumination from the top of the samples (Ominicure EXFO S2000). A two-step photopolymerization was carried out: monomeric coatings were first illuminated with UV light with designated intensities for 30 min to enable materials diffusion, then they were exposed to strong UV light (50 mW·cm⁻²) for 20 min to fully cure the monomers. A postcure was performed at 120 °C for 10 min. The final thickness of LCN coatings depends on the fingerprint pitch; the thickness of coating with pitch = 30 μm fingerprint was 4.5 μm.

2.3. Characterization. Surface topographies of LCN coatings were mapped with Digital Holographic Microscope (Lyncee Tech, Switzerland). The textures and alignment of samples were checked with microscope with crossed polarizers (Leica DM2700). The local mechanical properties of different domains in the LCN coating were studied with an atomic force microscope (AFM) with nanoindentation.40 The AFM measurements were carried out in the air in the PeakForce Quantitative Nanomechanical Mapping (QNM) mode by the Multimode 8 AFM with the NanoScope V controller (Bruker, USA). The cantilevers (OMCL-AC240TS, Olympus, Japan) had a nominal tip radius smaller than 8 nm, a resonance frequency of 70 kHz in air and spring constant 2.3 N/m, as measured by thermal fluctuations in air.41 The AFM piezo oscillated at 0.5 kHz and force-distance curves were captured each time the AFM tip tapped on the sample surface. The values of Young’s moduli were determined with reference to a material of a known Derjaguin–Müller–Toporov (DMT) elastic modulus by using so-called “relative method” based on a simplified DMT formalism. In the “relative method”, mechanical properties of a sample under investigation are compared to a known, well-defined polystyrene reference sample with Young’s modulus of 2.7 GPa (Bruker, USA). All AFM experiments were performed at room temperature of 20 °C with a stable humidity of 40%. The indentation depth (sample deformation) was controlled to be typically about 5 nm. The image processing and the data evaluation were performed with the NanoScope Analysis 1.9 software.

3. RESULTS AND DISCUSSION

3.1. Marangoni Effect in the Monomeric State and Anisotropic Polymerization Shrinkage. It has been demonstrated that low molecular weight chiral mixtures of cyanobiphenyl-based liquid crystals, when applied on a rubbed substrate with their other interface to air, form fingerprints textures with periodic corrugations in phase with the pitch of the planar chiral-nematic helix.42 Similar structures were found in fingerprint liquid crystal network coatings.42 The surface tension forces (Marangoni effect), eventually combined with the splay, bend and twist elastic forces within the liquid crystal, reduce the system’s free energy by forming spontaneous corrugations corresponding to the director periodicity. To demonstrate this effect for our LCNs, we made mixtures of nematic LC monomers 2–4 and chiral additive 5 and photoinitiator 1 or 7. Dissolved in THF the mixtures were
spin-coated on a substrate coated with a homeotropic aligning polyimide. After evaporation of the solvent, a film is obtained with a thickness of 4.5 \( \mu \)m. The fingerprint configuration is formed under the twisting force of monomer 5 and the homeotropic anchoring force of the polyimide layer. A typical texture of the still monomeric mixture as observed by polarizing optical microscope (POM) between crossed polarizers is shown in Figure 1b. By dynamic holographic microscopy a surface profile of the monomeric coating could be made (Figure 2a). By carefully correlating the POM with the DHM images of the same surface area, as is shown in Figure 2b, c and gratefully benefiting from the uniqueness of the fingerprints, we could determine that the molecular director in the hills and the valleys of the corrugations. At the top of the hills, the director is perpendicular toward the air surface (homeotropic following the liquid crystal nomenclature) and in the valleys the director is planar with the average long axes of the molecules parallel to the air surface. The confinement of the molecules in the helix configuration, align the molecules also perpendicular to the helix axes, which in our case also means parallel to the topographical ridges. The height of the corrugations in the monomeric state is around 30 to 50 nm (Figure 2a, e).

The surface topographical deformation during polymerization was in situ probed with DHM at room temperature (25 °C). During polymerization conducted in the presence of 2 wt % isotropic photoinitiator 7, the topography was observed to become inverted. When the polymerization has been completed, we find the planar domains are higher than the homeotropic domains. The height difference value now reaches to about 150–200 nm. The topographic inversion is confirmed by the correlated texture in POM image and the topography image (Figure 2c, d). The explanation can be found in polymerization shrinkage. Typically, the liquid crystal monomers, due to the conversion of van der Waals distances to covalent bond distances, exhibit a volume shrinkage of around 4–6% during polymerization, and simultaneously it was found that the shrinkage predominantly occurs along the director of the LC mixtures. In our system it means that the shrinkage in the homeotropic area is larger than in the planar area. In other words, the initially higher area shrinks more than the lower area and ultimately leads to the height inversion.

3.2. System I Coating: Polymerization Initiated by a Dichroic Photoinitiator. Dichroic species of anisotropic molecular shape have their large directional transition moment often parallel to their long molecular axes, thus largely interacting with the electrical field vector of light incoming along the normal to the coating surface. Consequently, when embedded in a fingerprint chiral-nematic matrix, dichroic molecules that follow the liquid crystal director have a large absorption when they are in the planar area of the molecular helix and a low absorption when they are in the perpendicular (homeotropic) area. Applying this dichroic optical property to the photoinitiator 1 promotes the polymerization to occur in the planar domains. Consuming the monomers faster in these regions stimulates concentration-gradient induced diffusion of the still unreacted monomer from the homeotropic area to the planar area. This will increase the volume of the planar area as no counter diffusion of the polymerized monomer takes place. Consequently, a relative expansion will take place in the \( z \)-direction perpendicular to the surface, resulting in the topographically higher planar domains.

![Figure 2](image_url)

*Figure 2.* Topography inversion in the polymerization process. (a) Monomeric state before polymerization. (b) After polymerization. The gray scale indicates the topographical height of the fingerprint patterns. (c) Correlated topography of the polymer LCN coating and (d) the corresponding POM image. (e) 2D topographical profiles.
To investigate the different light absorption of dichroic initiator in different domains, its dichroic behavior for the nonpolarized light was checked using a cholesteric sample and a homeotropic sample. The LC host for mixing the dichroic initiator consists of 10 wt % 2, 50 wt % 3, 40 wt % 4. For the cholesteric sample, additionally the chiral dopant 5 was added (1 wt % of LC host). From the UV–vis spectra (Figure 3a) of the samples still in their monomeric state, the dichroic ratio \( \frac{A_{\text{cholesteric}}}{A_{\text{homeotropic}}} \) of the dichroic photoinitiator 1 is 5.5 at 365 nm, indicating a good alignment of dichroic photoinitiator.

Figure 3. (a) UV–vis absorption spectra of dichroic initiator for nonpolarized light in cholesteric and homeotropic LC hosts. (b) Schematic illustration of the formation of system I coating. Because of material diffusion from homeotropic to planar domains and larger polymerization shrinkage in homeotropic domains, planar domains are higher than homeotropic domains. The white arrows represent the propagation direction of the light. (c) POM image between crossed polarizer. (d) Correlated surface topography image with the digital holographic microscope (DHM). The planar domains are higher than homeotropic domains. (e) 2D cross-section surface topography profile.

Figure 4. (a) Illustration of the quantification of the height difference between the planar and homeotropic domain. (b) Influence of dichroic initiator concentration on the height difference. Fixed parameters: light intensity 0.5 mW cm\(^{-2}\), temperature 30 °C. (c) Influence of light intensity on the height difference. Fixed parameters: 0.2 wt % dichroic initiator, polymerization temperature 30 °C. (d) Influence of polymerization temperature on the height difference. Fixed parameters: 0.2 wt % dichroic initiator; light intensity, 0.5 mW cm\(^{-2}\).
with the LC monomers. Following conventional free radical polymerization kinetics, the rate of free-radical photopolymerization scales with the root of the absorbed light intensity of the initiator. This is usually the case in the initial state of the polymerization. The difference in polymerization rate becomes even enhanced by the Trommsdorff effect at somewhat higher conversion where the termination reaction becomes suppressed by limited polymer chain mobility. When applied to our fingerprints with the helix axes parallel to the surface, the monomer conversion proceeds much faster in the planar domains than homeotropic domains and creates a periodic monomer concentration gradient following the helix contour, especially the fast consumed diacrylates. Consequently, LC monomers diffuse from homeotropic domains to planar domains as is schematically shown in Figure 3b.

The fingerprint LCN coatings we produced by means of the dichroic photoinitiator 1 we nominate as the system I coatings. In the monomeric state of the coatings the valleys are filled with the planar-aligned molecules. Polymerization with a nondichroic photoinitiator leads to height inversion and a corrugation depth of 150 nm. Now by polymerizing in the presence of the dichroic photoinitiator the height difference becomes even more pronounced as follows from the characterization of the same sample area using POM and DHM (Figure 3c-e). For clarity, we define \( h \) as the height difference between the planar and homeotropic domains (Figure 4a); its value is positive when planar domains are higher than the homeotropic domains. With slow polymerization rate to allow material diffusion at certain polymerization conditions (which will be discussed later in Figure 4), we can see that the nonbirefringent homeotropic domains are lower than the birefringent planar domains with a height difference \( h \) around 400 nm. Compared with the initial topography before polymerization where the homeotropic domains are 50 nm higher than planar domains, the relative height between homeotropic and planar domains changes by 450 nm in the polymerization process.

To gain better control over the height difference, the impact of dichroic initiator concentration, UV light intensity and polymerization temperature on the height difference between planar and homeotropic domains were investigated. In detail, on one hand, by varying the dichroic initiator concentration with keeping the UV light intensity constant (0.5 mW·cm\(^{-2}\)) or varying the UV light intensity while keeping the dichroic initiator concentration constant (0.2 wt %), the height difference is larger in the small initiator concentration or low light intensity regime, where the polymerization rate is relatively low and more materials diffusion is permitted (Figure 4b, c). On the other hand, when the initiator concentration is above 1 wt % or the light intensity is stronger than 5 mW·cm\(^{-2}\), no further change in height difference is observed with higher initiator concentration and light intensity. It is speculated that, at high initiator concentration and strong UV light intensity regime, the polymerization and cross-linking in homeotropic domains proceeds fast and inhibits the material diffusion from homeotropic to planar domains. At these conditions (large initiator concentration (>2 wt %) and strong UV light intensity (5 mW·cm\(^{-2}\)), the height difference is mostly contributed by the different polymerization shrinkage in homeotropic and planar domains. As for the influence of polymerization temperature, the height difference is observed to be larger at higher polymerization temperatures while other experimental parameters are kept constant (dichroic initiator concentration and UV light intensity are kept at 0.2 wt % and 0.5 mW·cm\(^{-2}\), respectively) (Figure 4d). In the experiment, we
selected temperature range to avoid crystallization and dewetting of the LC monomer coating.

3.3. System II Coating: Photopolymerization in the Presence of Dichroic dye. Next, we fabricated a different type of coatings (system II) by using the nondichroic photoinitiator 7 in the presence of dichroic dye 6. When the anisotropic light absorption is regulated by a dichroic dye, the photoinitiator will experience a lower UV light intensity in the planar regions. This slows down polymerization in planar domains and induces a concentration gradient-induced monomer diffusion and consequently the formation of elevated regions with homeotropic director orientation.

Similar to the investigation of the dichroic ratio of dichroic initiator 1 in section 3.2, the dichroism of dichroic dye 6 is revealed with a cholesteric sample and a homeotropic sample. The dichroic ratio ($A_{\text{cholesteric}}/A_{\text{homeotropic}}$) is calculated to be 4.0 at absorption peak 398 nm (Figure 5a), indicating that the dichroic dye exhibits stronger absorption to the light polarized parallel to its molecular long axis. Therefore, in fingerprint-configured LC coating, the dye has larger absorption of the normally incident light in planar domains than in the homeotropic domains. Consequently, the photoinitiator will experience an overall lower light intensity in the planar domains. This results in slower polymerization in the planar domains than in the homeotropic domains (Figure 5b). The faster conversion of LC monomers in the homeotropic domains induces diffusion to the homeotropic domains leading to local expansion. Although the polymerization shrinkage is larger in the homeotropic domains than in planar domains, the material diffusion dominates over the effect of polymerization shrinkage in this case.

Correlating the POM textures and the surface topography from DHM (Figure 5c–e) proves that the homeotropic domains are higher than the birefringent planar domains. The height difference $h$ is around $-200$ nm (coming from $-50$ nm before polymerization). To investigate the influencing factors on the height difference between homeotropic and planar domains in the prepared coating, several parameters including dichroic dye concentration, light intensity and polymerization temperature were studied. The results show that the height difference $h$ decreases monotonically with increased dichroic dye concentration, whereas all other factors remain the same (UV light intensity: 5 mW cm$^{-2}$, polymerization temperature: 30 °C) (Figure 6b).

Notably, a critical point of the dichroic dye concentration was found where $h=0$. In that case a flat coating surface is obtained. When the dichroic dye concentration (0.1–0.3 wt %) is lower than a certain threshold, the $h$ value is positive and homeotropic domains are lower than planar domains. This is attributed to a dominancy of the anisotropic polymerization shrinkage effect over the concentration gradient-induced material diffusion due to the relatively weak light absorption of the dichroic dye. Comparatively, with a higher concentration of dichroic dye, more light can be absorbed by dichroic dye molecules in planar domains, inducing slower polymerization initiation and more monomer diffusion from planar to homeotropic domains, resulting in smaller height difference $h$ value (Figure 6a).
As for the influence of UV intensity for the initiation of polymerization on the height difference, the evaluation was carried out with controlling the dichroic dye concentration (1.2 wt %) and a fixed polymerization temperature (30 °C). In the range from 0.2 to 40 mW cm\(^{-2}\), a maximum height difference was observed at 5 mW cm\(^{-2}\) (Figure 6b) which can be explained in terms of polymerization rate versus diffusion time. Within the nematic phase range, the larger height difference is observed at higher polymerization temperatures due to the higher diffusion rate of LC monomers.\(^{42}\) (Figure 6c) By varying the chiral dopant concentration in the monomer mixture, samples with different fingerprint pitches were fabricated. The height difference has a larger absolute value for samples with larger fingerprint pitches (Figure 6d) as more materials participate in the diffusion process.

As the surface corrugation is influenced by polymerization-induced diffusion of the various monomers to the high-intensity area, it is of interest whether the local chemical composition will change as well. In order to estimate this effect, especially the diffusion of the faster reacting diacrylate molecule, we measure the local moduli by AFM (Figure S1). For the samples without dye, the topographically higher planar domains were found to have a slightly lower modulus (2.4 GPa) than the homeotropic domains (2.9 GPa). Although in line with the expectation of a somewhat higher modulus along the director, this difference is relatively small. In contrast, in the system II coating where the same mixture is polymerized in the presence of the dichroic dye, the local modulus of planar domains (3.5 GPa) is 22% higher than that of the homeotropic domains (2.3 GPa). This indicates that the much smaller LC monomers diffuse faster to the high-intensity area, leaving a higher cross-link density in the lower planar domains.

### 3.4. Actuated Deformation of the Corrugated Fingerprints

We described earlier that fingerprint LCN coatings can be modulated by an in-plane electric field.\(^{22,43}\) The modulation typically proceeds on a time scale of several seconds. In relation to this work, it would be of interest how the fingerprint system I and system II coating would behave differently. As described earlier, the mechanism of the topographical surface deformation is based on the dynamic change of the scalar order parameter,\(^{19,22,44}\) because of which the homeotropic domains tend to shrink and planar domains tend to expand in the direction normal to the surface. As the initial surface landscapes of these coatings are different, we can afford different topographical deformation modes correspondingly. In detail, when actuated with the in-plane AC electric field which is generated with the interdigitated electrodes (Figure 1c), in system I coating, the height difference between the initially higher planar domains and the lower homeotropic domains becomes larger, making the surface more corrugated. The height difference increases from \(\sim 400\) nm to \(\sim 800\) nm when actuated with 10.2 V/\(\mu m\) AC electric field (900 kHz) (Figure 7a-d). Interestingly, system II coatings exhibit a different surface topographical response. With the homeotropic
domains being initially higher than planar domains, system II coating first transforms to a relatively flat surface when actuated with 5.1 V/μm AC electric field, and then adapts a corrugated surface with reversed surface topography when actuated with a larger electric field (10.2 V/μm) (Figure 7e–h). For the initially flat coatings which can be obtained by tuning the concentration of the dichroic dye and UV light intensity during polymerization (Figure S2), the flat surface turns into a corrugated state with h ~ 370 nm (Figure 7i–l). These different kinds of topographical deformation are promising for various potential applications, for instance for the dynamic removal of debris from surfaces or to affect the roughness-controlled wettability when actuated with the electric field (Figure S3).

4. CONCLUSIONS

In summary, we have shown methods to control the topographies of fingerprint LCN coatings using the dichroic dye and dichroic initiator. By controlling experimental parameters, we can tune the surface roughness with designated landscapes or with close to a flat surface. Moreover, we have control over the directionality of the molecular orientation in the hills and valleys of the corrugations which have prospects for adhesion/release properties of these surfaces and their surface wetting behavior. Corresponding experimental parameters influencing the height difference are systematically investigated, including the concentration of dichroic molecules, UV light intensity and polymerization temperature. Further we have demonstrated distinct topographical deformation modes of fingerprint LCN coating surfaces when actuated: system I coatings exhibit enhanced surface corrugations when actuated, whereas system II coating first undergoes a transition to a relatively flat surface and then affording an inverted topography when further actuated with larger electric field strength, and the coating with relatively flat surface becomes corrugated when actuated. We envision that this mask-free method to control surface topography can be expanded to stimuli-responsive liquid crystalline coatings with predesigned alignments beyond the fingerprint configuration. Moreover, the principle is universal for other coatings with domains of different molecular alignment, as obtained, for instance, by polymerization of LC monomers at substrates with patterned alignment layers.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsami.9b11928.

Flat surface mediated by material diffusion and polymerization shrinkage; contact angle change during electrical actuation; AFM topographic images and correlated modulus images measured with nanoindentation mode (PDF)

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Notes

The authors declare no competing financial interest.

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