Article

Synthesis of a Smart Conductive Block Copolymer Responsive to Heat and Near Infrared Light

Silvestre Bongiovanni Abel 1,2*, Kevin Riberi 1, Claudia R. Rivarola 1, Maria Molina 1,* and Cesar A. Barbero 1,*

1 Research Institute for Energy Technologies and Advanced Materials (IITEMA), National University of Río Cuarto (UNRC)-National Council of Scientific and Technical Research (CONICET), Ruta Nacional N° 36, Km 601, Río Cuarto (Córdoba) 5800, Argentina; bongiovanniabel.s@fi.mdp.edu.ar (S.B.A.);
kriberi@exa.unrc.edu.ar (K.R.);
crivarola@exa.unrc.edu.ar (C.R.R.)
2 Research Institute of Materials Science and Technology (INTEMA), National University of Mar del Plata (UNMdP)-National Council of Scientific and Technical Research (CONICET), Av. Colón 10850, B7608FDQ, Mar del Plata (Buenos Aires) 7600, Argentina
* Correspondence: mmolina@exa.unrc.edu.ar (M.M.); cbarbero@exa.unrc.edu.ar (C.A.B.);
Tel.: +54-358-467-6522 (M.M.); +54-9358-467-6157 (C.A.B.)

Received: 16 September 2019; Accepted: 18 October 2019; Published: 24 October 2019

Abstract: A method for the synthesis of a linear block copolymer (PNIPAM-b-PANI), containing a thermoresponsive block (poly(N-isopropylacrylamide), PNIPAM) and a Near Infrared (NIR) light-absorbing block (polyaniline, PANI), is reported. The synthetic approach involves a two-step successive polymerization reaction. First, the radical polymerization of NIPAM is done using 4-aminothiophenol as a chain transfer agent for the obtention of thermosensitive block terminated with an aniline (ANI) moiety. Second, the oxidative polymerization of ANI is initiated in ANI moiety of thermosensitive block to grow the second conductive PANI block. 1H nuclear magnetic resonance (NMR) and FT-IR spectroscopy shows the characteristics peaks of both polymeric blocks revealing the successful copolymerization process. Static Light Scattering (SLS) and UV-Visible combined measurements allowed the determination of the $M_w$ for PNIPAM-b-PANI macromolecule: $5.5 \times 10^5$ g mol$^{-1}$. The resulting copolymer is soluble in water (8.3 g L$^{-1}$) and in non-aqueous solvents, such as ethanol, formic acid, acetonitrile, and others. Both polymer blocks chains show the properties of the polymer chains. The block copolymer shows a lower critical solution temperature (LCST) at the same temperature (32–34 °C) than PNIPAM, while the copolymer shows pH dependent UV-vis-NIR absorption similar to PANI. The PNIPAM block suffers a coil to globule transition upon NIR light irradiation (785 nm, 100 mW), as shown by turbidimetry and Atomic Force Microscopy (AFM), due to local heating (more than 9 °C in 12 min) induced by the NIR absorption at the PANI block. Furthermore, the electrical conductivity of PNIPAM-b-PANI thin films is demonstrated (resistivity of $5.3 \times 10^{-4}$ Ω$^{-1}$ cm$^{-1}$), indicating that the PANI block is present in its conductive form.

Keywords: block copolymer; conductive polymer; thermal response; near infrared light; photothermal behavior; poly(N-isopropylacrylamide); polyaniline

1. Introduction

Conductive polymers (CPs) have attracted the attention of the scientific and technology community due to their electrochemical, electronic, and optical properties [1,2]. CPs have been also applied in nanomedicine [3], specifically as photothermal absorbers for anticancer and antibacterial photothermal therapy [4–6]. This is possible because CPs absorb electromagnetic radiation and transform it in heat, inducing tumor cell or bacteria death [7,8].
Polyaniline (PANI) is the most commonly used CP because of its thermal stability, low cost, and versatility [9]. However, the insolubility in common solvents and biological fluids is its main disadvantage [10]. Therefore, PANI could not reach the tumor cells to provoke the photothermal killing. Different strategies have been used to disperse or solubilize this CP. Some examples include chemical functionalization [11] or preparation of nanoparticles as stable dispersions in water [12,13]. We propose a new alternative, namely the block copolymerization of PANI with a water-soluble polymer chain.

Poly(N-isopropylacrylamide) (PNIPAM) is a well-studied water soluble and thermoresponsive polymer which presents a lower critical solution temperature (LCST) at ca. 32–34 °C [14].

In a previous work [15], PANI nanoparticles were stabilized with a water-soluble thermosensitive copolymer containing N-isopropylacrylamide (NIPAM) monomer units. The smart nanoparticles aggregated upon heating or NIR light irradiation by the local photothermal effect. Other kinds of applications were the combination of those polymers to generate macroscopic composites to be used as pressure sensors and thermal switch, microwave responsive devices, and chemo-mechanical actuators [16–18]. In recent years, a series of copolymers based on PNIPAM were synthesized, demonstrating the versatility of this thermosensitive polymer for the development of new materials [19–23]. Hsiao et al. [24] demonstrated that it is possible to grow PANI from the alkyl amine (R-NH₂) groups of chitosan producing another kind of NIR absorbing hydrogel.

In this work, a block copolymer (PNIPAM-b-PANI) made of a thermosensitive (PNIPAM) and a conductive (PANI) block is synthesized by a novel approach. First, PNIPAM polymer terminated with aniline moieties is generated by radical vinyl polymerization using an appropriate charge transfer agent (4-aminothiophenol, 4-ATF). In a second step, the PANI block is grown by oxidative polymerization of aniline (ANI), which initiates in the extreme of the PNIPAM block that contains the aniline group, in order to link the PNIPAM chains with PANI. The copolymer is soluble in aqueous solutions and some common solvents due to the effect of the soluble PNIPAM block. The obtained copolymer is characterized by FT-IR and NMR. The molecular weight determined by a combination of Static Light Scattering (SLS) and UV-visible spectroscopy.

The copolymer clearly aggregates in water when the temperature of the solution goes above 34 °C, showing that the thermosensitive behavior of the PNIPAM block is preserved. At the same time, the copolymer shows the typical UV-visible spectra of PANI, which presents a broad band at ca. 830 nm. Finally, when the solution is illuminated with NIR light (780 nm), the copolymer also aggregates. This behavior is possible because the PANI block absorbs the NIR light, locally heats the environment of the copolymer, and the phase transition of the PNIPAM block takes place when the LCST is overcome. In that way, the copolymer constitutes the smallest phase transition device induced photothermally.

2. Materials and Methods

2.1. Materials

N-isopropylacrylamide (NIPAM) used as monomer was obtained from Scientific Polymer Products (Wayne County, NY, USA), 2,2′-azobis(2-(2-imidazolin-2-yl)propane] dihydrochloride (VA044), ammonium peroxydisulfate (APS), and 4-aminothiophenol (4-ATF) were purchased from Sigma-Aldrich (St. Louis, MO, USA). HCl and methanol were obtained from Anedra. Anilinium hydrochloride (ANI) was purchased from Merck (Kenilworth, NJ, USA). Water used in this work was triply distilled. All reagents were analytical quality and used as received.

2.2. Synthesis of PNIPAM-b-PANI

The first synthetic step was carried out using a 1 mol L⁻¹ NIPAM solution in water and VA044 (5 × 10⁻³ g mL⁻¹) as thermally activated initiator agent of polymerization. Temperature (60 ± 0.1 °C), stirring conditions, and gas bubbling (N₂) were kept during 4 h to complete the polymerization process. 4-ATF was used as chain transfer agent at 3 × 10⁻⁵ mol L⁻¹ [25]. The aniline terminated PNIPAM molecule was purified by several heating-cooling cycles and centrifugation to remove remaining...
reactants or unreacted monomer. Then, the polymer was precipitated in methanol, obtaining the ATF terminated PNIPAM (PNIPAM-ATF) as a white solid.

Using the modified thermoresponsive block in water, the polymerization of ANI was carried out. 15 mL of 0.1 mol L\(^{-1}\) of ANI in 1 mol L\(^{-1}\) HCl (prepared and maintained at 0 ± 0.1 °C) was added to PNIPAM-ATF and stirred to obtain a clear solution. Then, APS (equimolar to ANI) was added as an oxidant agent to initiate the polymerization. The conducting polymer block grows from the terminal ANI groups at the modified PNIPAM molecules. In this way, a green solution of PNIPAM-b-PANI was obtained. Because PNIPAM is a vinylic chain polymer, no chain scission is expected by the effect of persulfate ions. Indeed, aniline polymerization inside PNIPAM gels is a well-established method, and no evidence of chain scission has been observed [26].

Finally, several heating–cooling cycles were carried out to purify the synthesized copolymer. In each cycle, the copolymer is aggregated by heating at 40 °C and separated by centrifugation. The supernatant discarded contain soluble reactants and side products, together with PNIPAM blocks not attached to PANI which suffers the coil to globule transition above the LCST but remains soluble, unlike the copolymer which contains the water insoluble PANI group. In that way, only block copolymer molecules are present in the solid. The purified precipitate is dried by freeze drying.

2.3. Solubility

Solubility of PNIPAM-b-PANI was tested at 20 °C in different solvents: 1-octanol (S. Aldrich, for High Performance Liquid Chromatography (HPLC) ≥ 99%), acetone (Aendra, 99.5%), acetonitrile (S. Aldrich, ACS Reagent ≥ 99.5%), carbon tetrachloride (S. Aldrich, anhydrous ≥ 99.5%), chloroform (Sintorgan, Buenos Aires, Argentina, HPLC grade), cyclohexane (Sintorgan, HPLC grade), cyclohexanol (S. Aldrich, ACS Reagent ≥ 99.5%), dichloromethane (S. Aldrich, anhydrous ≥ 99.8%), dimethyl sulfoxide (S. Aldrich, ACS Reagent ≥ 99.5%), ethanol (Sintorgan, HPLC grade), formic acid (Cicarelli, Pro-Analysis), isopropyl alcohol (S. Aldrich, ≥ 99.7%), N-methyl-2-pyrrolidone (S. Aldrich, ≥ 99.5%), pyridine (S. Aldrich, anhydrous ≥ 99.8%), tetrahydrofuran (S. Aldrich, anhydrous ≥ 99.9%), and water (Milli-Q quality, triply distilled). The solubility test was made by weighing a known mass (ca. 1 × 10^{-2} g) of the lyophilized copolymer and added to 2 mL of the chosen solvents. A Vortex device was used to help the solubilization process. By direct observation of the dissolution after 24 h, the samples were classified considering turbidity and precipitates as: non-soluble (no color in solution, colored precipitate presence), slightly soluble (cloudy solutions or particle presence), and soluble (clear colored solution, no precipitate presence).

2.4. Characterization Techniques

2.4.1. Molecular Weight \(M_w\) Determination

The \(M_w\) of the copolymer could not be determined by Gel Permeation Chromatography (GPC) and Static Light Scattering (SLS) due to the aggregation of PANI chains in usual GPC solvents. Therefore, a procedure was developed to determine the molecular weight \(M_w\) in two steps. First, the \(M_w\) of the PNIPAM block (terminated by aniline group) was determined by SLS measurements. A commercial SLS Malvern 4700 spectrometer (Malvern Instruments, Malvern, UK) equipped with a 488 nm laser OBIS was used. The details of instrumentation and theories have been described elsewhere. All measurements were carried out at 25 °C. The specific refractive index increments \((dn/dc)\) of PNIPAM in water at 25 °C were determined using a differential refractometer (BI-DCP, Brookhaven Instruments, New York, NY, USA). The measured \(dn/dc\) values of thermosensitive block at 25 °C and 480 nm in water was 0.1332 mL g\(^{-1}\). The mass of the conductive block in the copolymer was determined by UV-Visible spectroscopy measurements using a HP 8453 (Hewlett Packard, Palo Alto, CA, USA). The mass absorption coefficient of pure PANI (emeraldine base state) in N-methyl-2-pyrrolidone (NMP) was determined measuring the optical absorption at 328 nm. It is assumed that the absorption coefficient of the PANI block is the same.
The $M_w$ was calculated using the following equations:

$$m_{\text{copol}}^{\text{PNIPAM}} = m_{\text{copol}} - m_{\text{copol}}^{\text{PANI}}, \quad (1)$$

where $m_{\text{copol}}^{\text{PNIPAM}}$ is the mass of PNIPAM block, $m_{\text{copol}}$ is the mass of copolymer, and $m_{\text{copol}}^{\text{PANI}}$ is the mass of PANI block.

The number of PNIPAM chains was determined using the $M_w$ of the PNIPAM block and assuming 100% efficiency of the initiation by the aniline moieties of the PANI block formation; all PNIPAM chains have PANI chains attached, and the number of copolymer chains was calculated by Equation (2).

$$N_{\text{copol \ chains}} = N_{\text{PNIPAM \ chains}} = \frac{m_{\text{PNIPAM}}}{M_w^{\text{PNIPAM}}}. \quad (2)$$

In this way, the $M_w$ of the copolymer and the $M_w$ of the PANI block was calculated.

$$M_w^{\text{copol}} = \frac{m_{\text{copol}}}{N_{\text{copol \ chains}}}, \quad (3)$$

$$M_w^{\text{PANI}} = M_w^{\text{copol}} - M_w^{\text{PNIPAM}}. \quad (4)$$

### 2.4.2. Proton Nuclear Magnetic Resonance ($^1$H NMR)

The synthesized macromolecule was fully characterized by Proton Nuclear Magnetic Resonance ($^1$H NMR) in deuterated water. Samples were prepared drying the polymer in a vacuum device at 50 ± 0.1 °C during 48 h. Then, the copolymer was re-dissolved using deuterated water to analyze in the spectrometer. Proton nuclear magnetic resonance spectra were performed using a FT-NMR Bruker Ultra Shield 400 spectrometer (Bruker, Billerica, MA, USA) at 400 MHz $^1$H frequency.

### 2.4.3. Fourier Transformed Infrared Spectroscopy (FT-IR)

Fourier Transform Infrared Spectroscopy measurements were recorded in ATR mode on a FT-IR Perkin Elmer Model Spectrum Two (Perkin Elmer, Waltham, MA, USA. The device was equipped with a DynaScan interferometer. Software used as the interface was Spectrum 10. The spectrum was recorder from 600 to 2000 cm$^{-1}$ with a resolution of 4 cm$^{-1}$, setting 64 scans. Air was used as background, and the measurement was made at room temperature. The block copolymer sample (in conductive form, doped) was previously lyophilized and dried under vacuum at room temperature for 48 h.

### 2.4.4. UV-Visible Spectroscopy and Turbidimetry Measurements

A Hewlett-Packard 8453 diode array spectrophotometer was used to obtain UV-Visible spectra. Samples were diluted using triply distilled water. The spectra were taken in quartz cells (0.1 cm path length). The spectra of the doped and undoped copolymer were measured using buffers to control pH of the sample (doped: pH 4; undoped: pH 10). The wavelength range for measurement was 300–900 nm. Turbidimetry was determined by measuring the absorption at a wavelength of 600 nm in quartz cells. Temperature was set up in the range between 25–42 °C by heating in water baths.

### 2.4.5. Conductivity Test

The resistance measurement was made on an interdigital copper electrode with small gap (width = 400 µm) between fingers. A PNIPAM-$b$-PANI drop was deposited on the electrode. The resistance
was quantified as time function using a DC multimeter and a Bs20x Data Recording System V4.x as the interface program. The thickness of the film was estimated from the mass of copolymer deposited. The area of the deposit was measured from the calibrated photograph, and the density of the copolymer was determined with a pycnometer.

2.4.6. Photothermal Effect

In testing the photothermally induced effect, free standing solution drops (5 × 10^{-3} g L^{-1}) of PNIPAM-b-PANI in water were irradiated by a NIR laser (780 nm and 100 mW power) through a fiber optic. Temperature of the drops was monitored by a TES 1326S/1327K infrared thermometer calibrated with a type K thermocouple. Duration of experiment was 15 min, recording the temperature at one-minute intervals, as previously described [15]. A thermographic camera Ti10 IR-Fusion Technology (Fluke, Everett, WA, USA) was used to capture IR images. Digitalization was made with a SmartView® software.

2.4.7. Atomic Force Microscopy (AFM)

AFM has been used successfully to study the morphology of block copolymers film [27–30]. Here, we used to monitor the reversible changes of microstructure induced by the triggering of a coil to globule transition of PNIPAM blocks. An atomic force Agilent 5420 AFM/STM microscope (Agilent, Santa Clara, CA, USA) was used to take measurements of thin films before and after irradiate the macromolecule using the NIR laser. Drops (ca. 5 × 10^{-3} g L^{-1}) were deposited on flat mica surfaces, and dried in air before scanning. A Point Probe® Plus Non-Contact/Tapping Mode-Long Cantilever (PPP NCL) with a force constant of 6 N m^{-1} and resonance frequency of 156 Hz was used in the Acoustic AC (AAC) mode. Images were taken before and after two min of NIR irradiation.

3. Results and Discussion

The block copolymer based on PNIPAM and PANI was synthesized in two steps (Figure 1). First, a PNIPAM chain terminated in aniline unit was obtained by free radical polymerization. In this sense, the NIPAM (1) monomer is used to polymerize using an azo initiator (2) in the presence of a chain transfer agent (4-ATF) (3) to yield PNIPAM chains terminated by aniline groups (4, PNIPAM-ATF) [25]. This reaction is followed by the oxidative polymerization of aniline (5) monomer with APS as oxidant. The PANI chain grows from the –S–C_{6}H_{4}–NH_{2} extreme of PNIPAM block. This is a well-known method to grow PANI chains attached to other polymer chains [31] or surfaces [32]. In that way, the linear block copolymer (PNIPAM-b-PANI) (6) is obtained.

PNIPAM chains are soluble in water and do not absorb in the UV-visible range, while PANI chains are insoluble and strongly colored (green in acid media). The copolymer obtained as a green water-soluble powder. The solubility in water results 8.3 g L^{-1}. Photographs of each block (PNIPAM and PANI) and the resulting material (PNIPAM-b-PANI) are shown in Figure 2.

The copolymer structure was studied by 1H NMR (Figure 3). 1H NMR of PNIPAM-ATF shows following peaks: δ: 1.13 (s, 6H, isopropyl groups of NIPAM), 1.57 (2H, polymer backbone), 2.00 (1H, polymer backbone), and 3.70–4.00 (1H, NIPAM). The results are in agreement with previous study of PNIPAM using NMR [33]. All the aforementioned peaks and a new signal at ~7.7 ppm are observed [13]. 1H NMR PNIPAM-b-PANI: δ: 1.13 (s, 6H, isopropyl groups of NIPAM), 1.57 (2H, polymer backbone), 2.00 (1H, polymer backbone), 3.70–4.00 (1H NIPAM), and 7.50–8.00 (4H, aromatic protons of PANI).
The copolymer structure was studied by NMR (Figure 3). The following peaks were observed: δ: 1.13 (s, 6H, isopropyl groups of NIPAM), 1.57 (2H, polymer backbone), 2.00 (1H, polymer backbone), 3.70–4.00 (1H, NIPAM), and 7.50–8.00 (4H, aromatic protons of PANI). The results are in agreement with previous studies. All the mentioned peaks and a new signal at ~7.7 ppm are observed. The copolymer preparation can be controlled by the monomer feed ratio.

The broad signal due to aromatic protons in PANI is quite weak. It is likely that the rigid nature of the PANI chains broadens the signal and decreases the intensity [34]. Therefore, it cannot be used to calculate the copolymer composition but reveals that aniline monomer units are present in the copolymer.
IPAM and PANI are soluble. Since only solubility was determined by several authors for the vibrational mode of the C–C stretching. The transmittance decreases upon an increase of temperature with an increase of temperature (Figure 5a). The band at 1140 cm\(^{-1}\) is in agreement with the reported by several authors for the vibrational mode of protonated amines (–N–H\(^{+}\)) generated during the acid doping process of the CP \([35,36]\); (v) the presence of the N–H bending corresponding to the amide II of PNIPAM is detected at 1530 cm\(^{-1}\); and (vi) the C=O stretching of amide I of the thermosensitive block appears at 1625 cm\(^{-1}\) \([37]\).

The FT-IR spectrum for PNIPAM-b-PANI is shown in Figure 4a. After a thorough analysis of the main bands, the assignation is following detailed: (i) the presence of band at ca. 1410 cm\(^{-1}\) corresponds to C=C; (ii) the band corresponding to the C–N stretching is present at 1298 cm\(^{-1}\); (iii) the bands between 750–900 cm\(^{-1}\) are referred to C–H of Ar present in the PANI backbone; (iv) the prominent band at 1140 cm\(^{-1}\) is in agreement with the reported by several authors for the vibrational mode of protonated amines (–N–H\(^{+}\)) generated during the acid doping process of the CP \([35,36]\); (v) the presence of the N–H bending corresponding to the amide II of PNIPAM is detected at 1530 cm\(^{-1}\); and (vi) the C=O stretching of amide I of the thermosensitive block appears at 1625 cm\(^{-1}\) \([37]\).

The UV-Visible spectra show the electronic transitions of the PANI block (Figure 4b) which change upon exposure to acid or basic media in agreement with previous results for PANI nanostructures, such as films, nanofibers, and nanospheres \([38,39]\). As expected, the pH sensitivity of the PANI block is unaffected by the presence of the PNIPAM block. This result is relevant for photothermal therapy.
applications since the spectra show absorption in the NIR range (>750 nm) [40–42]. Besides, it allows the building of sensors/actuators devices that respond to UV-Visible and NIR spectral range.

The $M_w$ was determined in two steps, as described before. First, the $M_w$ of the PNIPAM block was determined by SLS measurements, giving a value of $3.8 \times 10^5$ g mol$^{-1}$ (degree of polymerization 3360). In a second step, the $M_w$ of the conductive block was calculated to be $1.7 \times 10^5$ g mol$^{-1}$ (degree of polymerization = 1870). This value is in good agreement with the reported ones for the molecular weight of chemically synthesized PANI measured by GPC by Wei and coworkers [43]. Adding these values, we could estimate that the block copolymer has a $M_w$ of $5.5 \times 10^5$ g mol$^{-1}$.

The block copolymer turned out to be soluble in several aqueous and organic solvents where PNIPAM is soluble but PANI is insoluble (Table 1). Obviously, the copolymer is soluble in those solvents marked with # in Table 1 where both PNIPAM and PANI are soluble. Since only solubility in water, where PNIPAM shows a clear coil to globule transition, is required to achieve the goal of the paper, no attempt was made to increase the solubility.

| Solvent                | Solubility  |
|------------------------|-------------|
| 1-octanol              | Insoluble   |
| Acetone                | Soluble     |
| Acetonitrile           | Soluble     |
| Carbon tetrachloride   | Insoluble   |
| Chloroform             | Insoluble   |
| Cyclohexane            | Insoluble   |
| Cyclohexanol           | Slightly soluble |
| Dichloromethane        | Insoluble   |
| Dimethyl sulfoxide     | Soluble     |
| Ethanol                | Soluble     |
| Formic acid #          | Soluble     |
| Isopropyl alcohol      | Soluble     |
| N-methyl-2-pyrrolidone #| Soluble     |
| Pyridine               | Soluble     |
| Tetrahydrofuran        | Slightly Soluble |
| Water                  | Soluble     |

*A PANI is soluble in these solvents.

The turbidity of a PNIPAM-b-PANI aqueous solution is studied as a function of the increasing temperature (Figure 5a). The transmittance decreases upon an increase of temperature with an inflection point at ~34 ºC. This value is close to the LCST of PNIPAM [19]. At lower temperatures, the PNIPAM block is extended in the solution interacting by hydrogen bonding with water, making the copolymer soluble. At temperatures higher than LCST, the PNIPAM block becomes globular decreasing its interaction with water and making the copolymer insoluble. Therefore, the solution became cloudy, decreasing the transmittance due to increased dispersion. This change upon heating confirms the presence of a PNIPAM block covalently linked to the PANI block. It is important to remark the total reversibility of the behavior when the room temperature is newly achieved. This fact could allow the fabrication of a thermal switch at the molecular level. The photograph inserted in Figure 5a confirms the observed changes for thermally induced polymer aggregation.
The high solubility of the block copolymer in common solvents allows obtaining thin films PNIPAM-b-PANI by depositing drops of its solution, followed by solvent evaporation. By measurement of the electrical resistance of the films deposited on an interdigitated electrode and estimating the film thickness from the deposited mass, the resistivity value for the molecular nanocomposite is found to be of ca. $5.3 \times 10^{-4} \ \Omega^{-1} \ \text{cm}^{-1}$. The value is significantly lower than that of pure PANI ($\sim 1 \ \Omega^{-1} \ \text{cm}^{-1}$) [38,39] but it is in the order of PANI statistical copolymers with substituted anilines ($10^{-3} - 10^{-4} \ \Omega^{-1} \ \text{cm}^{-1}$) [44,45] or blends with dielectric polymers [46]. The microstructure of blends should be similar to that of the copolymers’ films. Indeed, a blend of 5% PANI in PVA showed conductivities in the same order than the block copolymer [47]. Moreover, the conductivity could be increased by changing the deposition procedure. Since it was not the goal of the work, no attempt was made to optimize the conductivity. However, the conductivity values measured are only two orders of magnitude lower than carefully optimized CP loaded PNIPAM based membranes [48]. The result suggests that the PANI blocks are electrically connected in the solid state, but forming a composite while PNIPAM blocks act as an insulating matrix.

Figure 5. Material characterization of PNIPAM-b-PANI. (a) Turbidimetry as temperature function in aqueous solution; (b) photothermal effect on drop of PNIPAM-b-PANI aqueous solution under Near Infrared (NIR) irradiation; (c) optical and 3D infrared photograph of an aqueous drop containing the copolymer before and after NIR irradiation.
The morphological changes, upon NIR irradiation, of deposited thin films on a mica surface were studied by Atomic Force Microscopy (AFM) (Figure 6). A clear change of topography is observed after 2 min of NIR irradiation since the smooth surface of the deposited copolymer becomes populated by nanometrically sized globules. The NIR absorbing block (PANI) absorbs light and heat up by non-radiative decay. When the temperature overcomes the LCST of the thermosensitive (PNIPAM) block, inducing the coil to globule transition of PNIPAM forming arrays of globular shaped aggregates on the surface. The transition face of PNIPAM block can be observed in the film because it contains some retained water. A typical line profile (Figure 6, right) clearly shows the topographical changes upon irradiation. The small undulations (with a mean value of 0.03 nm) in the non-irradiated copolymer film (upper profile) become dominated by larger features (ca. 0.2 nm of size) upon NIR irradiation.

This effect could be useful for the application of the material in photothermally responsive surface or smart coatings with control of the hydrophobicity properties by remote irradiation [49].

![Figure 6. Atomic Force Microscopy (AFM) images of PNIPAM-b-PANI thin films: (a) before and (b) after NIR irradiation.](image)

4. Conclusions

The results demonstrate that a copolymer containing a thermosensitive and a conjugated/conductive block induce a phase transition by the photothermal effect of light absorbed by the conjugated block.

The thermoresponsive and conductive block copolymer can be obtained by a simple synthetic method involving two different successive polymerization reactions. The structure of the copolymer is ascertained by FTIR and NMR spectroscopy. The molecular weight and mean composition of the copolymer is evaluated using a combination of SLS and UV-visible spectroscopy.

PNIPAM-b-PANI turned out to be soluble in a wide range of organic solvents and water, unlike PANI. It was demonstrated that the block copolymer maintain the I properties of both blocks: the thermosensitivity of PNIPAM (with a LCST ~34 °C) and the electronic absorption/conductivity of PANI (directly dependent on the pH). Moreover, upon NIR irradiation, the PANI block absorbs the light and heat up, triggering the phase transition of the PNIPAM block. Such a transition provokes a morphologic change of the copolymer deposited as film and is detected by AFM. The controlled change at a distance using NIR light makes the copolymer a promising photothermal therapy agent since NIR light could penetrate several centimeters in biological tissue.

Thin films obtained by deposition show electrical conductivity, revealing the presence of active PANI moieties.
Besides the interesting properties of PNIPAM-b-PANI, the synthetic method could be applied to produce other block copolymers where either the conductive block (e.g., pH sensitive aminobenzoic acids) [50] or the vinylic block (e.g., pH sensitive polyacrylic acid) [51] have different properties. All the mentioned properties make the synthesized smart copolymer promising for diverse applications (e.g., photothermal devices, surfaces and coatings, and resistive sensors).

Supplementary Materials: The following are available online at http://www.mdpi.com/2073-4360/11/11/1744/s1, Figure S1: Optical photographs of a copolymer solution drop under NIR at different irradiation times.

Author Contributions: Conceptualization, C.A.B., C.R.R. and M.M.; methodology, M.M and C.R.R.; investigation, K.R., M.M. and S.B.A.; resources, C.A.B.; writing—original draft preparation, S.B.A. and M.M.; writing—review and editing, S.B.A., C.A.B., C.R.R. and M.M.; visualization, S.B.A.; supervision, M.M. and C.A.B.; project administration, C.A.B.; funding acquisition, C.A.B.

Funding: This research was funded by FONCYT, PICT 1760/2016, CONICET, PDTS-CIN IP693.

Acknowledgments: Bongiovanni Abel and Riberi thank CONICET for a graduate study fellowship. Rivarola, Molina, and Barbero are permanent research fellows of CONICET.

Conflicts of Interest: The authors declare no conflict of interest.

References
1. Ansari, M.O.; Ansari, S.A.; Cho, M.H.; Ansari, S.P.; Abdel-wahab, M.S.; Alshahrie, A. Conducting Polymer Nanocomposites as Gas Sensors. In Functional Polymers; Springer: Berlin, Germany, 2019; pp. 911–940.
2. Zarrintaj, P.; Khalili, R.; Vahabi, H.; Saeb, M.R.; Ganjali, M.R.; Mozafari, M. Polyaniline/metal oxides nanocomposites. In Fundamentals and Emerging Applications of Polyaniline; Elsevier: Amsterdam, The Netherlands, 2019; pp. 131–141.
3. Kašpářková, V.; Humpholiček, P.; Stejskal, J.; Capáková, Z.; Bober, P.; Skopalová, K.; Lehocký, M. Exploring the Critical Factors Limiting Polyaniline Biocompatibility. Polymers 2019, 11, 362. [CrossRef]
4. Vines, J.B.; Lim, D.-J.; Park, H. Contemporary Polymer-Based Nanoparticle Systems for Photothermal Therapy. Polymers 2018, 10, 1357. [CrossRef]
5. Xu, L.; Cheng, L.; Wang, C.; Peng, R.; Liu, Z. Conjugated polymers for photothermal therapy of cancer. Polym. Chem. 2014, 5, 1573–1580. [CrossRef]
6. Zarrintaj, P.; Vahabi, H.; Saeb, M.R.; Mozafari, M. Application of polyaniline and its derivatives. In Fundamentals and Emerging Applications of Polyaniline; Elsevier: Amsterdam, The Netherlands, 2019; pp. 259–272.
7. Liao, G.; Li, Q.; Xu, Z. The chemical modification of polyaniline with enhanced properties: A review. Prog. Org. Coat. 2019, 126, 35–43. [CrossRef]
8. Acevedo, D.F.; Rivarola, C.R.; Miras, M.C.; Barbero, C.A. Effect of chemical functionalization on the electrochemical properties of conducting polymers. Modification of polyaniline by diazonium ion coupling and subsequent reductive degradation. Electrochim. Acta 2011, 56, 3468–3473. [CrossRef]
9. Morrin, A.; Ngamma, O.; O’Malley, E.; Kent, N.; Moulton, S.E.; Wallace, G.G.; Smyth, M.R.; Killard, A.J. The fabrication and characterization of inkjet-printed polyaniline nanoparticle films. Electrochim. Acta 2008, 53, 5092–5099. [CrossRef]
10. Chen, F.; Liu, P. Conducting polyaniline nanoparticles and their dispersion for waterborne corrosion protection coatings. ACS Appl. Mater. Interfaces 2011, 3, 2694–2702. [CrossRef]
14. García-Peñas, A.; Biswas, C.S.; Liang, W.; Wang, Y.; Yang, P.; Stadler, F.J. Effect of Hydrophobic Interactions on Lower Critical Solution Temperature for Poly(N-isopropylacrylamide-co-dopamine Methacrylamide) Copolymers. *Polymers* 2019, 11, 991. [CrossRef]

15. Bongiovanni Abel, S.; Molina, M.A.; Rivarola, C.R.; Kogan, M.J.; Barbero, C.A. Smart polyaniline nanoparticles with thermal and photothermal sensitivity. *Nanotechnology* 2014, 25, 495602. [CrossRef]

16. Rivero, R.E.; Molina, M.A.; Rivarola, C.R.; Barbero, C.A. Pressure and microwave sensors/actuators based on smart hydrogel/conductive polymer nanocomposite. *Sens. Actuators B* 2014, 190, 270–278. [CrossRef]

17. Martinez, M.V.; Bongiovanni Abel, S.; Rivero, R.; Miras, M.; Rivarola, C.; Barbero, C. Polymeric nanocomposites made of a conductive polymer and a thermosensitive hydrogel: Strong effect of the preparation procedure on the properties. *Polymers* 2015, 78, 94–103. [CrossRef]

18. Molina, M.; Rivarola, C.; Miras, M.; Lescano, D.; Barbero, C. Nanocomposite synthesis by absorption of nanoparticles into macroporous hydrogels. Building a chemomechanical actuator driven by electromagnetic radiation. *Nanotechnology* 2011, 22, 245504. [CrossRef]

19. Tu, X.; Meng, C.; Liu, Z.; Sun, L.; Zhang, X.; Zhang, M.; Wei, H. Synthesis and phase transition of poly(N-isopropylacrylamide)-based thermo-sensitive cyclic brush polymer. *Polymers* 2017, 9, 301. [CrossRef]

20. Li, J.; Yang, L.; Fan, X.; Wang, F.; Zhang, J.; Wang, Z. Multi-Responsive Behaviors of Copolymers Bearing N-Isopropylacrylamide with or without Phenylboronic Acid in Aqueous Solution. *Polymers* 2018, 10, 293. [CrossRef]

21. Rwei, S.P.; Chiang, W.Y.; Way, T.F.; Tuan, H.; Chang, Y.C. Study of the thermo-pH-Sensitivity of Stereo-Controlled Poly(N-isopropylacrylamide-co-IAM) Copolymers via RAFT Polymerization. *Polymers* 2018, 10, 512. [CrossRef]

22. Grimm, O.; Schacher, F. Dual Stimuli-Responsive P (NIPAam-co-SPA) Copolymers: Synthesis and Response in Solution and in Films. *Polymers* 2018, 10, 645. [CrossRef]

23. Yoneoka, S.; Park, K.; Nakagawa, Y.; Ebara, M.; Tsukahara, T. Synthesis and Evaluation of Thermoresponsive Boron-Containing Poly(N-isopropylacrylamide) Diblock Copolymers for Self-Assembling Nanomicellar Boron Carriers. *Polymers* 2019, 11, 42. [CrossRef]

24. Hsiao, C.-W.; Chen, H.-L.; Liao, Z.-X.; Sureshbabu, R.; Hsiao, H.-C.; Lin, S.-J.; Chang, Y.; Sung, H.-W. Effective Photothermal Killing of Pathogenic Bacteria by Using Spatially Tunable Colloidal Gels with Nano-Localized Heating Sources. *Adv. Funct. Mater.* 2015, 25, 721–728. [CrossRef]

25. Valdebenito, A.; Encinas, M.V. Thiophenols as chain transfer agents in the polymerization of vinyl monomers. *Polymer* 2005, 46, 10658–10662. [CrossRef]

26. Molina, M.A.; Rivarola, C.R.; Barbero, C.A. Effect of copolymerization and semi-interpenetration with conducting polyanilines on the physicochemical properties of poly(N-isopropylacrylamide) based thermosensitive hydrogels. *Eur. Polym. J.* 2011, 47, 1977–1984. [CrossRef]

27. Rasmont, A.; Leclère, P.; Doneux, C.; Lambin, G.; Tong, J.D.; Jérôme, R.; Lazzaroni, R. Microphase separation at the surface of block copolymers, as studied with atomic force microscopy. *Colloids Surf. B* 2000, 19, 381–395. [CrossRef]

28. Leclère, P.; Lazzaroni, R.; Brédas, J.L.; Yu, J.M.; Dubois, P.; Jérôme, R. Microdomain morphology analysis of block copolymers by atomic force microscopy with phase detection imaging. *Langmuir* 1996, 12, 4317–4320.

29. Raybin, J.; Ren, J.; Chen, X.; Gronheid, R.; Nealey, P.F.; Sibener, S.J. Real-time atomic force microscopy imaging of block copolymer directed self assembly. *Nano Lett.* 2017, 17, 7717–7723. [CrossRef]

30. Zhang, X.; Douglas, J.F.; Jones, R.L. Influence of film casting method on block copolymer ordering in thin films. *Soft Matter* 2012, 8, 4980–4987. [CrossRef]

31. Hosseini, S.H.; Musavi, S.M. Preparation and Characterization of Conducting Polystyrene Graft Polyaniline. *Life Sci. J.* 2012, 9, 4912–4925.

32. Rubinstein, I.; Rishpon, J.; Sabatani, E.; Redondo, A.; Gottfeld, S. Morphology control in electrochemically grown conducting polymer films. Precoating the metal substrate with an organic monolayer. *J. Am. Chem. Soc.* 1990, 112, 6135–6136. [CrossRef]

33. Ohta, H.; Ando, I.; Fujishige, S.; Kubota, K. A 13C PST/MAS NMR study of poly(N-isopropylacrylamide) in solution and in the gel phase. *J. Mol. Struct.* 1991, 245, 391–397. [CrossRef]

34. Nechtschein, M.; Santier, C.; Travers, J.P.; Chroboczek, J.; Alix, A.; Ripert, M. Water effects in polyaniline: NMR and transport properties. *Synth. Met.* 1987, 18, 311–316. [CrossRef]
35. Rodrigues, P.C.; Cantão, M.P.; Janissek, P.; Scarpa, P.C.; Mathias, A.L.; Ramos, L.P.; Gomes, M.A. Polyaniline/lignin blends: FTIR, MEV and electrochemical characterization. Eur. Polym. J. 2002, 38, 2213–2217. [CrossRef]

36. Trchová, M.; Šeděnková, I.; Stejskal, J. In-situ polymerized polyaniline films. FTIR spectroscopic study of aniline polymerisation. Synth. Met. 2005, 154, 1–4. [CrossRef]

37. Broglia, M.F.; Balmaceda, I.; Carrizo, F.; Barbero, C.; Rivarola, C.R. Acid hydrogel matrixes as reducing/stabilizing agent for the in-situ synthesis of Ag-nanocomposites by UV irradiation: pH effect. Mater. Res. Express 2019, 6, 055021. [CrossRef]

38. Huang, W.S.; MacDiarmid, A.G. Optical properties of polyaniline. Polymer 1993, 34, 1833–1845. [CrossRef]

39. Stejskal, J.; Trchová, M.; Bober, P.; Humpolíček, P.; Kašpárková, V.; Sapurina, I.; Shishov, M.A.; Varga, M. Conducting Polymers: Polyaniline. In Encyclopedia of Polymer Science and Technology; John Wiley & Sons: Hoboken, NJ, USA, 2002.

40. Cheng, L.; Zhang, F.; Wang, S.; Pan, X.; Han, S.; Liu, S.; Yuan, Q. Activation of Prodrugs by NIR-Triggered Release of Exogenous Enzymes for Locoregional Chemo-photothermal Therapy. Angew. Chem. Int. Ed. 2019, 58, 7728–7732. [CrossRef]

41. Zou, H.; Tang, D.; Wang, N.; Jia, S.; Sun, Z.; Yang, X.; Peng, J. Polyethylene glycol–modified molybdenum oxide as NIR photothermal agent and its ablation ability for HeLa cells. Colloid Polym. Sci. 2019, 297, 249–260. [CrossRef]

42. Liu, C.; Zhang, S.; Li, J.; Wei, J.; Müllen, K.; Yin, M. A Water-Soluble, NIR-Absorbing Quaterrylenediimide Chromophore for Photoacoustic Imaging and Efficient Photothermal Cancer Therapy. Angew. Chem. Int. Ed. 2019, 58, 1638–1642. [CrossRef]

43. Tang, X.; Sun, Y.; Wei, Y. Molecular weight of chemically polymerized polyaniline. Die Makromol. Chem. Rapid Commun. 1988, 9, 829–834. [CrossRef]

44. Huang, J.; Kaner, R.B. A general chemical route to polyaniline nanofibers. J. Am. Chem. Soc. 2004, 126, 851–855. [CrossRef]

45. Bongiovanni Abel, S.; Olejník, R.; Rivarola, C.R.; Slobodian, P.; Sáha, P.; Acevedo, D.F.; Barbero, C.A. Resistive sensors for organic vapors based on nanostructured and chemically modified polyanilines. IEEE Sens. J. 2018, 18, 6510–6516. [CrossRef]

46. Razak, S.I.A.; Rahman, W.A.W.A.; Hashim, S.; Yahya, M.Y. Polyaniline and their conductive polymer blends: A short review. Malays. J. Fundam. Appl. Sci. 2013, 9. [CrossRef]

47. Dutta, P.; Biswas, S.; Ghosh, M.; De, S.K.; Chatterjee, S. The dc and ac conductivity of polyaniline–polyvinyl alcohol blends. Synth. Met. 2001, 122, 455–461. [CrossRef]

48. Kim, H.; Kim, K.; Lee, S.J. Nature-inspired thermo-responsive multifunctional membrane adaptively hybridized with PNIPAm and PPy. NPG Asia Mater. 2017, 9, e445. [CrossRef]

49. Molina, M.A.; Rivarola, C.R.; Broglia, M.F.; Acevedo, D.F.; Barbero, C.A. Smart surfaces: Reversible switching of a polymeric hydrogel topography. Soft Matter 2012, 8, 307–310. [CrossRef]

50. Salavagione, H.J.; Acevedo, D.F.; Miras, M.C.; Motheo, A.J.; Barbero, C.A. Comparative study of 2-amino and 3-aminobenzoic acid copolymerization with aniline synthesis and copolymer properties. J. Polym. Sci. Part A 2004, 42, 5587–5599. [CrossRef]

51. Luützenkirchen, J.; van Male, J.; Leermakers, F.; Sjöberg, S. Comparison of Various Models to Describe the Charge–pH Dependence of Poly(acrylic acid). J. Chem. Eng. Data 2011, 56, 1602–1612. [CrossRef]