**Article**

**Eco-Friendly Disposable WS₂ Paper Sensor for Sub-ppm NO₂ Detection at Room Temperature**

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**Abstract:** We developed inexpensive and disposable gas sensors with a low environmental footprint. This approach is based on a biodegradable substrate, paper, and features safe and nontoxic electronic materials. We show that abrasion-induced deposited WS₂ nanoplatelets on paper can be employed as a successful sensing layer to develop high-sensitivity and selective sensors, which operate even at room temperature. Its performance is investigated, at room temperature, against NO₂ exposure, finding that the electrical resistance of the device drops dramatically upon NO₂ adsorption, decreasing by ~42% (~31% half a year later) for 0.8 ppm concentration, and establishing a detection limit around ~2 ppb (~3 ppb half a year later). The sensor is highly selective towards NO₂ gas with respect to the interferents NH₃ and CO, whose responses were only 1.8% (obtained for 30 ppm) and 1.5% (obtained for 8 ppm), respectively. Interestingly, an improved response of the developed sensor under humid conditions was observed (tested for 25% relative humidity at 23 °C). The high-performance, in conjunction with its small dimensions, low cost, operation at room temperature, and the possibility of using it as a portable system, makes this sensor a promising candidate for continuous monitoring of NO₂ on-site.

**Keywords:** WS₂; humidity sensor; nitrogen dioxide; paper-based device; room temperature sensor; 2D materials

**1. Introduction**

Gas sensing is becoming more and more important in our society. In fact, detection of various gases in low concentrations is crucial (and sometimes even mandatory) in fields such as air quality assessment, greenhouse gas emissions control, the quantification of volatiles for smart maintenance in the industry sector, and identification of biomarkers in medical diagnosis [1–5].

Among the gaseous species that should be monitored, detection of nitrogen dioxide (NO₂) is required in different applications. In the atmosphere, NO₂ plays the role of greenhouse gas and causes acid rain and photochemical smog problems [6]. Long-term exposure to high levels of NO₂ produces harmful effects for humans and other living beings [7,8]. Additionally, nitrogen oxides (NOₓ) in exhaled breath are biomarkers for inflammatory and oxidative changes in lungs, serving as early indicators of the pathophysiology of many respiratory diseases [9].

Chemical sensors based on semiconductor materials [10–15], and particularly metal oxides, are the most popular devices to sense NO₂ gas [16–19]. However, these metal-oxide-based devices present poor sensitivity at room temperature, requiring high-temperature...
operation that leads to high power consumption and the eventual degradation of the sensing material [20]. Moreover, metal-oxide devices require the use of substrates compatible with micro-fabrication techniques (i.e., silicon, glass, quartz, etc.), which hampers their application in disposable electronics applications where the use of ultra-low cost and biodegradable substrates is crucial to emerging technologies and environmental impacts.

Over the past two decades, thanks to the revival of interest in van der Waals materials aroused by the isolation of graphene [21], sensors based on layered materials have been presented as a real step forward in gas sensing. Their exceptionally large surface-area-to-volume ratio makes these materials strongly sensitive to adsorbed gases, and therefore they are promising candidates for gas detection [22–28]. In fact, over the last years, several examples of NO$_2$ gas sensors based on van der Waals materials, operating even at room temperature, have been proved [26,29–32].

The attractive properties of the conventional printer paper as a substrate, mainly its environmental-friendliness and low-cost, have led researchers to develop paper-based devices for various applications, including memory devices [33], solar cells [34,35], RFID-enabled wireless sensors [36], or supercapacitors [37]. Recently, some of the authors have demonstrated the integration of van der Waals materials on paper substrates through direct abrasion against the rough surface of paper [38–41]. However, only light and temperature sensors have been demonstrated so far, with gas sensing remaining unexplored. Because of the combination of ultra-low cost, availability, and biodegradability of paper substrates, integrating van der Waals materials on paper substrates opens the door for low-cost and disposable [42–50] gas sensors.

Here, we demonstrate the fabrication of gas sensors on standard copy paper substrates using abrasion-induced deposited WS$_2$ films as a sensing material. This process is simple to implement and yields low-cost and environmentally-friendly devices. In fact, standard copy paper substrates are biodegradable, and the sensing film (WS$_2$) and electrodes (graphite) are safe, nontoxic materials that can be found as natural minerals on Earth’s crust. The sensing performance of the WS$_2$-based sensor under exposure to NO$_2$ gas, operating at room temperature, is examined. Furthermore, the selectivity relative to potential interfering gases (NH$_3$ and CO) is analyzed [51–53].

2. Materials and Methods

2.1. Materials

Standard (untreated) copy printer paper (80 g/m$^2$) was used as supporting substrates because of its low cost and availability. Tungsten disulfide (WS$_2$) from HAGEN automation Ltd. (Bedford, UK) (0.6 microns APS Ultra Grade Micronized) was used as gas sensing channel material. Among the different semiconducting transition metal dichalcogenides, we selected WS$_2$ as it yielded films with the lower electrical resistance facilitating the electrical read-out of the fabricated devices. Graphite pencil (Madrid, Spain) (4B, Faber Castell) was employed to pattern graphite-based electrical leads (it has $\sim$80% of graphite content [54]) to connect the WS$_2$ channel to the readout electronics.

2.2. Sensor Fabrication

The steps for the gas sensor fabrication are depicted in Figure 1. First, the outline of the sensitive layer channel and electrodes were printed on the paper substrate (Figure 1a). Then, a stencil mask (made of Nitto SPV 224 tape) delimited the sensitive area (Figure 1b). Micronized WS$_2$ powder was rubbed against the unmasked paper substrate with a cotton swab (Figure 1c). The depositing process mimics the action of drawing/writing with a pencil on paper, where the friction forces between the van der Waals materials and paper cleaves the van der Waals crystals, leading to a network of interconnected platelets. The powder was abraded until a continuous film was reached. Then, the excess powder and the stencil mask were removed (Figure 1d). In the last process step, graphite electrodes were deposited on top of the sensitive material by drawing directly with a high-graphite content
pencil (Figure 1e). These electrodes were contacted with spring-loaded probes (pogo pins) integrated inside the test chamber. Figure 1f shows a picture of the final device.

![Process sequence for the sensor fabrication. (a) The outline of the device is printed with a standard office printer using standard copy paper. (b) Nitto SPV 224 tape is used to mask around the active area of the device. (c) An active film of WS$_2$ nanosheetlets is deposited by mechanical abrasion of micronized WS$_2$ powder against the paper surface. (d) The mask is removed, showing the patterned WS$_2$ film on paper active area of the device. (e) Graphite electrodes are patterned by simply filling in, with a 4B graphite pencil, the area between the device outline dotted lines. (f) Picture of the final fabricated device.](image)

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2.3. Material Characterization

Scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDX), using a FE-SEM, FEI Nova NANOSEM 230 (Hillsboro, OR, USA), were used to characterize the morphology and the composition of the WS$_2$ films deposited on paper. An electron energy of 7 keV was employed for imaging and 14 keV for EDX spectroscopy.

2.4. Experimental Setup

The chemoresistive sensor was placed inside a 6.25 mL volume airtight chamber for its characterization in different reducing and oxidizing atmospheres. Airflow inside the chamber was set to 100 mL/min, switching between gas sample for 10 min (exposition time) and synthetic air for 20 min (purge time). Gas cylinders supplied target gases with appropriate concentrations and balanced with the carrier gas (synthetic air): NO$_2$ (1 ppm), CO (10 ppm), and NH$_3$ (50 ppm) (all of them from Nippon Gases). Then, the initial sample concentration was diluted with synthetic air by using a gas mixing unit (GMU, Ray IE, Cáceres, Spain) to obtain the required exposed concentration. For proper control of the relative humidity (RH) inside the chamber, a handheld thermohygrometer RS1364 was used. The temperature was kept at 23 °C during the tests, and the required RH was achieved with a third flow controller that regulates the synthetic air bubbling through deionized water (Figure 2).

The sensor was kept at room temperature while variations of the resistance over time were recorded with a digital multimeter (Keithley 2001). The experiment control and real time data acquisition was implemented with a PC using an in-house custom-made software developed with LabVIEW. The response of the sensor was calculated with the following equation:

$$\text{Response} = \frac{\Delta R}{R_0} = \frac{R - R_0}{R_0}$$

(1)

where $R$ is the electrical resistance for the sensor in the tested gas and $R_0$ is the resistance of the sensor in the air.
Figure 2. Experimental setup used to measure different gases concentrations with the WS$_2$-on-paper sensor in real time.

3. Results

3.1. Structural and Morphological Characterization

Figure 3a shows SEM image of the cross-section of the WS$_2$ flakes forming the micronized WS$_2$ powder. The flakes are initially 5–10 µm in lateral size and 50–150 nm thick. Figure 3b shows a SEM image of the porous microscopic structure of the WS$_2$ film deposited onto the paper substrate, formed by interconnected crystalline WS$_2$ platelets, ensuring a very large effective surface area of the device. During the abrasion process, the WS$_2$ flakes are cleaved, reducing their lateral dimensions to 1–5 µm and their thickness to sub-50 nm. Figure 3c shows a low magnification SEM image of a WS$_2$ film obtained after its deposition on the paper substrate. The bare paper has fibrous-like structures arising from the cellulose fibers. The abrasion-induced deposition method yielded a continuous film of packed WS$_2$ platelets covering the fibers. The bare paper and WS$_2$ film can be easily distinguished because of their different contrast under SEM inspection. Figure 3d shows a SEM image of the sensitive area/electrode interface where it can be observed a sizable change in contrast due to the difference in electrical conductivity between the WS$_2$ film and the WS$_2$ film covered with graphite. The chemical composition of the film was characterized by energy dispersive X-ray (EDX) spectroscopy. Apart from the prominent W and S peaks, expected from the WS$_2$ film, the spectrum had peaks associated with the presence of C and O, arising from the paper substrate. The spectrum also showed a Ca peak, attributed to the presence of calcium carbonate, a filler usually added to paper pulp to achieve a brighter white color (Figure 3e).

3.2. Electrical Characterization

A thorough characterization of the electrical properties of abrasion-induced deposited WS$_2$ films on copy paper can be found in Ref [41]. Briefly, the resistivity of the films, determined through current vs. voltage measurements in transfer length configuration, ranges from ~360 Ω·m to ~530 Ω·m and electric field effect measurements demonstrated the p-type character of the WS$_2$ film. Additionally, in the mentioned reference, 118 devices were developed to study the reproducibility, showing a low dispersion taking into account the nature of the films: a random network of interconnected platelets where percolation transport is expected.

The sensor was kept at air atmosphere, and after a few minutes, the calculated root-mean square (RMS) noise level was approximately 0.01% for the sensor. Thanks to the electrical continuity of the WS$_2$ film, the device operates with low noise that is a consequence of the dry deposition method, and a good adhesion between sensitive material and paper fibers.
Figure 3. SEM images of (a) the cross-section of the WS\textsubscript{2} flakes deposited on paper by abrasion technique, (b) the network of WS\textsubscript{2} flakes deposited on paper after abrasion-induced deposition, (c) the interface between bare paper and the deposited WS\textsubscript{2} film, and (d) the border between the surface of the WS\textsubscript{2} sensing area and the WS\textsubscript{2} covered by the graphite electrode. (e) EDX spectrum for the micronized WS\textsubscript{2} deposited on the paper.

3.3. Gas Sensor Characterization

To characterize the performance of this sensor, its sensitivity, and response time, we studied the changes in resistance upon cyclic exposition and purge processes with NO\textsubscript{2} at various concentrations ranges (0.2 ppm–0.8 ppm, see Figure 4a). The gas sensing mechanism is attributed to the surface reactions between the p-type WS\textsubscript{2} platelets and gas molecules. In the case of a p-type semiconductor in an oxidant environment (NO\textsubscript{2}), the concentration of electrons on the surface decreases (the number of holes increases) and, consequently, the resistance of the WS\textsubscript{2} film decreases (Figure 4a). The sensor device showed a fast recovery with a low baseline drift of 0.6% at 0.8 ppm of NO\textsubscript{2}. Therefore, an automatic baseline subtraction method based on linear correction for measurements before exposition and in the final of the purge time was implemented.

In most real-life applications, the target gas is in a complex environment surrounded by several gases at different concentrations, requiring sensors with high sensitivity and selectivity to discriminate and classify the target gas. Important interfering gases, in the above applications, are carbon monoxide (CO) and ammonia (NH\textsubscript{3}) [55,56]. Therefore, gas sensors with negligible interference between reducing and oxidizing environments, i.e., a high absolute selectivity, are highly desirable to achieve a more reliable signal interpretation. To test the selectivity of the WS\textsubscript{2} on paper NO\textsubscript{2} sensor, we have subjected the device to cyclic exposition and purge processes with CO and NH\textsubscript{3} at various concentrations ranges (1.5 ppm–8 ppm for CO and 10 ppm–30 ppm for NH\textsubscript{3}, see Figure 4b,c). Upon exposure to CO and NH\textsubscript{3}, the resistance increases as expected for a p-type semiconductor, because the generated electrons recombine with holes. The gas test showed the sensor has a remarkably
higher sensitivity towards NO$_2$ (42% resistance change at 0.8 ppm) than NH$_3$ and CO, whose responses were 1.8% (obtained at 30 ppm) and 1.5% (obtained at 8 ppm), respectively (see Figure 5a,b). This can be justified by the adsorption kinetics of gas molecules on the sensitive material. Additionally, the high sensitivity and selectivity to NO$_2$ is consistent with results of density functional theory calculation in Ref. [32]. Interestingly, this article explains that the chemically reactive edge sites of WS$_2$ served as highly favorable active sites for direct interaction with target NO$_2$ gas molecules. This is consistent with the fact that abrasion-induced is an effective method to generate numerous edge sites in deposited WS$_2$ nanoplatelets on paper, since the technique induces fracturing, tearing, and peeling off from substrates.

![Figure 4](image)

**Figure 4.** Paper-based sensor response to cyclic exposure to (a) NO$_2$, (b) NH$_3$, and (c) CO gases.

Accordingly, the response speed of the device was studied. We determined the response time parameter $\tau_{90}$, defined as the time necessary to reach approximately 90% of the response when the sensor is subjected to an abrupt change in atmosphere. The $\tau_{90}$ values obtained were NO$_2$ 5.2 min at 0.8 ppm, NH$_3$ 8.8 min at 30 ppm, and CO 9.6 min at 8 ppm (Figure 5c). In comparison, the paper-based sensor showed the shortest response time for NO$_2$ that nearly achieved the equilibrium. In contrast, CO and NH$_3$ responses had not yet approached an equilibrium, resulting in a high sensitivity NO$_2$ gas sensor with
The improved response to NO₂ under humid conditions (25% RH at 23°C with 25% RH and it responded efficiently to humidity, obtaining a maximum response of 114% (Figure 7a). Cross-sensitivity measurements were carried out to 0.8 ppm of NO₂ to the paper-based sensor to 0.8 ppm of NO₂ (red), 30 ppm of NH₃ (green), and 8 ppm of CO (blue).

Figure 5. (a) Real time responses, (b) maximum (absolute value) responses and (c) response time, τ₉₀, of the paper-based sensor to 0.8 ppm of NO₂ (red), 30 ppm of NH₃ (green), and 8 ppm of CO (blue).

The structural continuity of the micronized WS₂ particles deposited by abrasion provides higher electrical conductivity toward a lower limit of detection (LOD) compared to sensors fabricated by other methods, such as drop-casting [26]. From the response for 0.8 ppm of NO₂, a theoretically achievable LOD of around 2 ppb was calculated, which is equivalent to a signal-to-noise ratio (SNR) value of three.

In order to assess the stability of these devices upon environmental degradation, we performed a new set of measurements half a year after its fabrication (the sensor was stored under ambient conditions during that time). Thereafter, the response for 0.8 ppm of NO₂ was slightly decreased to 31%, increasing the LOD around 3 ppb (Figure 6). In particular, the decrease of the gas response is small between measurements for a half year interval, which demonstrates that the paper-based sensor has a slow degradation, maintaining a high response over time. The effect of relative humidity on the paper-based sensor was tested at 23°C with 25% RH and it responded efficiently to humidity, obtaining a maximum response of 114% (Figure 7a). Cross-sensitivity measurements were carried out to assess the influence of RH on the sensor response to NO₂. Figure 7b illustrates the effect of 0.8 ppm of NO₂ detection in an environment with a RH of 25% with a sensor response of ~44%.

The experimental responses of NO₂ over time and under humid conditions were compared for 0.8 ppm (Figure 8a1–a4). In the days following device manufacture, the sensor had a high response close to 42% with an operating resistance of ~4 MΩ. Then, after half a year where the sensor was stored in ambient conditions, the resistance increased to ~22 MΩ, decreasing the response to ~31%, which was attributed to the effects of sensor poisoning by gases surrounding in ambient during the half a year period. However, a positive effect of humid conditions (25% RH at 23°C) is that at the same gas concentration the sensor response increased to 44%, simultaneously the resistance scaled up ~142 MΩ. The improved response to NO₂ with humidity can be justified by the intrinsic and induced dipole moments of the molecules and their intermolecular charge transfer [57]. Finally, there was a practically total regeneration of the sensor after humidity exposition was obtained for dry synthetic air, and the sensor showed a response of ~30% for a resistance operation of ~42 MΩ (Figure 8b,c). Therefore, the sensor performance is a huge benefit since it can work on a large range of tests for multidisciplinary applications carried out in humid conditions with sensitivity gain.
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Figure 6. Real time response of the sensor obtained with NO$_2$ sample at the concentration of 0.8 ppm for the as-fabricated device (red) and after half a year of exposure to ambient conditions (black).

The experimental responses of NO$_2$ over time and under humid conditions were compared for 0.8 ppm (Figure 8a–d). In the days following device manufacture, the sensor had a high response close to 42% with an operating resistance of ~4 MΩ. Then, after half a year where the sensor was stored in ambient conditions, the resistance increased to ~22 MΩ, decreasing the response to ~31%, which was attributed to the effects of sensor poisoning by gases surrounding in ambient during the half a year period. However, a positive effect of humid conditions (25% RH at 23 °C) is that at the same gas concentration the sensor response increased to 44%, simultaneously the resistance scaled up ~142 MΩ. The improved response to NO$_2$ with humidity can be justified by the intrinsic and induced

Figure 7. Real time response of the paper-based sensor tested at 23 °C (a) with intervals of dry air (0% RH) and 25% RH (b) for 0.8 ppm of NO$_2$ in an environment with RH 25% at 23 °C.
positive effect of humid conditions (25% RH at 23 °C) is that at the same gas concentration there was a practically total regeneration of the sensor after humidity exposition was observed. The improved response to NO$_2$ gas, which makes it a promising candidate for monitoring of NO$_2$ sensing.

In summary, we fabricated and characterized a disposable NO$_2$ sensor based on a p-type WS$_2$ film on standard paper. The sensing film was deposited by a low-cost and easy to implement abrasion-induced method, establishing a nanostructured sensitive layer by exfoliation of micronized WS$_2$ particles and an electrical connection among flakes. The structure of the WS$_2$ sensing film was characterized by using SEM, which showed rough and porous film formed by interconnected WS$_2$ flakes. The sensor showed excellent sensing properties at room temperature with a response higher than 42% (31% half a year later) at 0.8 ppm NO$_2$ and with a significant LOD of around 2 ppb (3 ppb half a year later). The relative humidity of 25% at 23 °C has a beneficial impact. The result indicates the high sensitivity, selectivity, and repeatability of the presented sensor towards sub-ppm level of NO$_2$ gas, which makes it a promising candidate for monitoring of NO$_2$ sensing.

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