Visible light photoluminescence in TiO$_2$/CdS nanopowders synthesized by sol-gel route: effect of gel aging time

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A series of sol-gel TiO$_2$/CdS, TiO$_2$ powders and coagulated CdS nanoparticles were studied by XRD, HRTEM and Raman spectroscopy to elucidate the effect of low-temperature gel aging time on visible photoluminescence (PL) emission of the TiO$_2$/CdS composites. With an increase in aging time a content of amorphous titania and incorporated CdS nanoparticles decreases in composites. For all composites, visible PL emission includes bands attributed to surface oxygen vacancies and hydroxyl group of TiO$_2$ nanocrystals, as well as yellow-green and red bands related to lattice defect states of CdS nanoparticles. It was found that gel aging time is a crucial parameter to influence visible PL emission in composites. This emission is suppressed with increasing aging time, and its evolution shows that healing of oxygen vacancy defects and hydroxyl group affect visible emission more significantly than improving crystallinity degree. The correlation between visible PL emission in TiO$_2$/CdS and their visible-light photocatalytic activity was discussed.

Keywords: TiO$_2$/CdS composites, nanoparticles, photoluminescence, defect states, sol-gel route.

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1. Introduction

Semiconductor TiO$_2$/CdS composites represent an important group of visible-light photoactive materials and are attracting considerable attention due to the fact that cadmium sulfide has a relatively narrow band gap and is well combined with the band structure of titanium dioxide [1–3]. The development of physical and chemical methods for the synthesis of nanoscale semiconductor particles and quantum dots has provided new possibilities for the creation of photoactive heterogeneous structures based on these phases. In nanosized systems, a more efficient carrier transfer takes place owing to the improved interface contacts [4, 5]. The efficiency of the photon energy conversion into photocurrent and the photocatalytic activity depend not only on a charge carrier transfer rate across the interface of the TiO$_2$/CdS couple but also on a recombination rate of the photogenerated electron-hole pair within titanium dioxide and CdS layers [2].

For visible-light photoactive nano-sized TiO$_2$/CdS composites [6–8], as well as for the visible photoactive TiO$_2$-doped materials [9,10], the direct correlation has been often found between the increased photoactivity and decreased radiative recombination in the visible range. Visible-light emission, related to the radiative recombination in nanosized TiO$_2$ is mainly attributed to (I) the self-trapped excitons, (II) oxygen vacancies and (III) surface states [11–14]. For CdS nanocrystals and nanoparticles, visible PL includes a number of emission bands associated with both surface electron states and states of complex defect centers [15–18]. Visible PL emission is affected not only by surface and lattice defects of TiO$_2$ nanocrystals but their morphology [13], and as well as by the crystallinity degree of titania [19]. For TiO$_2$/CdS composites, recombination rate has been found to increase with increasing degree of defects in TiO$_2$ layers [20] and imperfection of TiO$_2$/CdS interface [5]. Concentration ratio of TiO$_2$/CdS is also an essential parameter which affects the visible PL emission, with the concentration dependence of PL emission being non-monotonic [6].

Currently, several synthesis strategies are being actively developed to obtain visible photoactive TiO$_2$/CdS nanocomposites, and by varying the synthesis conditions it is possible to change their structural properties, defect states and electron structure for tuning PL emission. In one of them, pre-synthesized CdS nanoparticles are directly incorporated into TiO$_2$ matrix during titania formation process via sol-gel route [21–24]. As a rule, the influence of the initial concentrations of the sensitizing additive on the emission and photocatalytic properties of sol-gel TiO$_2$/CdS composites is studied [21]. At the same time, when preparing TiO$_2$/CdS composites, the sol-gel route is sometimes used with low-temperature heating of the sol-gel [23,24], the effect of which on PL emission was not monitored. Recently, a new
The aim of work is to elucidate the effect of gel aging time on visible PL emission in TiO$_2$/CdS composites prepared by sol-gel technique using pre-synthesized aqueous colloidal solution of Cd$_2$S nanoparticles. The details of TiO$_2$/CdS preparation can be found in [24]. In brief, a certain amount of titanium (IV) n-butoxide was dissolved in the aqueous colloidal Cd$_2$S solution. Sol-gel samples were heated up to the boiling point under continuous stirring and aged at this temperature for 1, 3 and 4 h. To remove the remaining organic species, the powders were centrifuged and dried in air at 120$^\circ$C for 3 h. Bare TiO$_2$ powders were also prepared by the same hydrolysis route in the pure deionized water. The aging time at boiling point was 1 and 4 h.

X-ray diffraction (XRD), high resolution transmission electron microscopy (HRTEM) and Raman micro-spectroscopy in resonance and off-resonance modes were employed to study structural state of TiO$_2$/CdS composites. The XRD patterns of samples were recorded by a Shimadzu MAXima-X XRD-7000 (Shimadzu, Japan) automatic diffractometer with CuK$_\alpha$ radiation ($\lambda = 1.5406$ Å) in 2$\theta$ angle range 10–80$^\circ$ with a step 0.03$^\circ$ and an exposure time of 10 sec at each step. The volume fractions and structural characteristics of constituent phases were calculated employing PCW 2.4 software [27]. High resolution TEM images and electron diffraction patterns were obtained with the help of a JEM-2100 microscope (JEOL, Japan) equipped with an Energy Dispersive X-ray Analyzer (EDX). HRTEM images were processed with the Digital Micrograph software package. Raman and photoluminescence spectra were excited by a low-power (up to 10 mW) laser irradiation at wave length of 480, 514 and 633 nm at room temperature, and they were recorded by a LabRAM HR800 (Horiba, Japan) spectrometer, providing a focal spot on the samples of 1 – 2 $\mu$m diameter. The spectral resolution was about 1 cm$^{-1}$. Raman and photoluminescence spectra were analyzed with multi-peak Gaussian fitting method using the “Peakfit v 4.11” software package.

2. Experimental part

TiO$_2$/CdS composites were prepared by direct hydrolysis technique using pre-synthesized aqueous colloidal solution of Cd$_2$S nanoparticles. The details of TiO$_2$/CdS preparation can be found in [24]. In brief, a certain amount of titanium (IV) n-butoxide was dissolved in the aqueous colloidal Cd$_2$S solution. Sol-gel samples were heated up to the boiling point under continuous stirring and aged at this temperature for 1, 3 and 4 h. To remove the remaining organic species, the powders were centrifuged and dried in air at 120$^\circ$C for 3 h. Bare TiO$_2$ powders were also prepared by the same hydrolysis route in the pure deionized water. The aging time at boiling point was 1 and 4 h.

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aging, respectively. This effect was discussed in detail using the Molecular Dynamic simulations in our previous work [26]. The drop in the concentration of CdS nanoparticles is associated with the decreases in the content of the amorphous phase in the TiO$_2$ matrix and the calculated thermodynamic stability of the composite CdS @ TiO$_2$ nanoparticles depending on the structure of the titanium shell: amorphous titania > polycrystalline titania (anatase or brookite). HRTEM images confirmed that colloidal CdS nanoparticles are incorporated into all anatase/brookite matrices [26]. A typical HRTEM image of the incorporated CdS nanoparticle with zone axis [001] is given in the Fig. 1(b).

![HRTEM image of a CdS nanoparticle incorporated into the TiO$_2$ matrix, [001] zone axis. The filtered image is presented in the inset (b)](image)

**Fig. 1.** X-ray diffraction patterns for the sol-gel TiO$_2$/CdS composites synthesized with varying gel aging time. The lines (1), (2) and (3) correspond to 1, 3 and 4 hours of aging (a). HRTEM image of a CdS nanoparticle incorporated into the TiO$_2$ matrix, [001] zone axis. The filtered image is presented in the inset (b)

A series of the micro-Raman spectra recorded from the different regions of powders demonstrates the good uniformity of structural properties of all synthesized samples TiO$_2$/CdS. Fig. 2 presents the typical untreated and treated Raman spectra for the TiO$_2$/CdS powders recorded with wavelength excitation equal to 633 nm. The most intense peaks in the treated spectra belong to titania phases (anatase and brookite). All 6 active modes of anatase phase (149 $(E_{1g})$, 200 $(E_{2g})$, 640 $(E_{1g})$, 401 $(B_{1g(1)}$)), 517 cm$^{-1}$ $(A_{1g}$ and $B_{1g(2)}$)) can be detected for powders. There are weaker peaks which can be attributed to the modes of brookite phase [30]. These peaks are 128 $(A_{1g})$, 166 $(A_{1g})$, 218 $(B_{1u})$, 245 $(A_{1g})$, 323 $(B_{1g})$, 366 $(B_{2g})$ and 622 $(B_{1u})$ cm$^{-1}$. It can be seen that the corresponding peaks are shifted to the higher frequencies and smeared as compared to that of standard microcrystalline TiO$_2$-anatase, indicating that the size effect for nanocrystals takes place. For all TiO$_2$/CdS powders, including powder with CdS concentration about 1%, in Raman spectra excited at 633 nm (1.95 eV), there is a smeared band with maximum around 300–302 cm$^{-1}$ which becomes stronger for laser wavelength of 514 nm at resonance condition and can be uniquely assigned to the first longitudinal optical (LO) mode of CdS [31].

XRD, HRTEM and Raman spectroscopy revealed that all TiO$_2$/CdS composite powders have an anatase / brookite matrix with a phase ratio close to 2:1 and a small content of amorphous titanium dioxide. With an increase in the gel aging time at boiling point, the content of the amorphous component decreases from 15 to 5%, and the average crystallite size of the anatase and brookite phases increases slightly from 6.5 to 8 nm. The CdS particles incorporated into the TiO$_2$ matrix have an average size of approximately 5-7 nm, which is close to the average particle size in the initial CdS colloidal solution. The structure of the incorporated particles, as in the initial solution, is characterized by a random hexagonal close-packed lattice.

### 3.2. Photoluminescence properties of TiO$_2$/CdS composites

The inset in Fig. 2(a) presents the untreated Raman spectra recorded with a 633 nm excitation wavelength. As can be seen, with a decrease of gel aging time the untreated spectra demonstrate consistently increasing background due to visible light luminescence. In order to investigate the effects of luminescence and their aging time dependence in more detail, Raman spectra were recorded with a 480 nm excitation wavelength for TiO$_2$/CdS composites, CdS nanoparticles coagulated from the colloidal initial solution and bare TiO$_2$ powder synthesized by the same sol-gel
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![Image]

**FIG. 2.** Raman spectra for the sol-gel TiO$_2$/CdS (1, 2, 3), bare TiO$_2$ (4) nanopowders and microcrystalline anatase (5), excitation is 633 nm. The peaks of anatase (A) are signed. The peaks of CdS and brookite are marked with triangles and upside-down triangles, respectively. Notations (1, 2, 3) are the same as on Fig. 1(a). The untreated spectra are shown in insert route with varying gel aging time. Deconvolution of the peaks was performed to gain more insight into the nature of defect states which determine the observed luminescence in different samples.

At first, consider the PL spectra of the bare TiO$_2$ nanopowder and CdS nanoparticles. Fig. 3(a,b) demonstrate visible light and NIR (near infrared) PL emission for the bare TiO$_2$ nanopowder synthesized with gel aging time of 1 and 4 h. For both sample, broad visible PL emission can be fitted with fore Gaussian sub-band centered at 533 nm (2.3 eV), 580 nm (2.1 nm), 633 nm (1.95 nm) and 670 nm (1.85 eV). One band at 850 nm (1.46 eV) is observed in NIR region. The all visible emission bands can be associated to the radiative recombination at the surface defects sites such as oxygen vacancy and hydroxyl groups which are the dominant trapped sites for nano-powders fabricated by low-temperature sol-gel. One set visible emissions at 2.3 and 1.95 eV are attributed to the deexcitation from the lower levels of oxygen vacancies associated with Ti$^{3+}$ in anatase lattice to the ground state [12, 14]. The sub-band at 2.1 eV is due to the deexcitation from lower levels in Ti$^{3+}$ 3d states of TiO$_2$ lattice to the deep levels created by OH$^-$ group [12]. The observed NIR PL emission at 1.46 eV is a signature of the brookite phase and appears due to the radiative recombination at the intrinsic lattice defects acting as trapped sites [32]. As a consequence of the fabrication process (sol-gel route), we can suppose that visible emission appears owing to the surface oxygen vacancies in brookite nanoparticles, and one additional sub-band centered at 1.85 nm is attributed to these defects.

For bare TiO$_2$ nanopowders, the intensity of all sub-bands of the visible light PL emission noticeably decreases with the increasing gel aging time at boiling point. Therefore, the changes in the surface defect concentrations in TiO$_2$ nano-powder with gel aging time significantly affect the visible PL emission, unlike the phase transformation of amorphous TiO$_2$ into a crystalline state which proceeds during the aging stage. In the latter case, the intensity of PL emission should have increased with the increases in the content of the crystalline phase [19].

A PL emission spectrum for CdS nanoparticles coagulated from the colloidal solution is presented in Fig. 3(c). A broad sub-band centered at 540 nm (2.3 eV) and a shoulder at 710 nm (1.75 eV) extended up to 940 nm (1.3 eV) are revealed. A similar PL band structure was observed earlier for CdS nanopowder which exhibits a structure related to that of CdS coagulated nanoparticles and is characterized by a high concentration of lattice stacking faults [17]. The yellow-green emission at 540 nm can be endorsed to the deexcitation from the Cd interstitial states to the valence band [15,17,18], and the red emission shoulder is believed to be caused by transitions of electrons trapped in the surface states to the valence band and to be increased with the accumulation of crystallographic defects in CdS structure [18].

Figure 3(d–f) demonstrates PL emission spectra for TiO$_2$/CdS composites fabricated by sol-gel using pre-synthesized CdS colloidal nanoparticles and with varying gel aging. The PL spectra of TiO$_2$/CdS composites managed to
Fig. 3. PL emission for sol-gel TiO$_2$ (a,b), coagulated CdS nanoparticles (c) and sol-gel TiO$_2$/CdS nanopowders (d-f), excitation is 480 nm. For TiO$_2$, gel aging time is 1 and 4 hours. For TiO$_2$/CdS, notations (1, 2, 3) are the same as on Fig. 1(a).
be decomposed by the same sub-bands as in bare TiO$_2$ nanopowder and CdS particles, with sub-band positions being changed slightly. The main difference in the spectra recorded for TiO$_2$/CdS powders with 1, 3 and 4 hours of gel aging stage is the decrease in the intensity of all emission sub-bands related to both the TiO$_2$ matrix and the incorporated CdS nanoparticles, with increasing gel aging time. As in the case of pure TiO$_2$, the effect of the phase transformation from amorphous TiO$_2$ to crystalline phases, which occurs during gel aging, on the emission of TiO$_2$/CdS composite powders is insignificant. In samples with short aging time (1 h) and containing the maximum amount of CdS nanoparticles up to 9%, visible emission is maximum, and is significantly higher than in the original pure components. As shown above and discussed in detail in our previous work [26], there are no significant differences between the structural properties of the components of the TiO$_2$/CdS composite and the original bare sol-gel TiO$_2$ nano-powder and CdS nanoparticles. However, the increased defect concentrations in the TiO$_2$/CdS interface region, detected by MD simulations of CdS @ TiO$_2$ composite particles, will affect surface defect states and lead to an increased visible PL emission, especially in the early stages of aging. With an increase in the aging time to 4 hours, PL emission drops significantly due to the decrease in the content of incorporated CdS nanoparticles to 1% and due to the healing of TiO$_2$/CdS interface defects and surface defects of TiO$_2$ nanocrystallites as in the case of the bare TiO$_2$ nanopowder.

Consider the relation between the observed visible PL emission for TiO$_2$/CdS powders synthesized with varying gel aging time and their visible light photocatalytic activity, previously tested in the oxidative degradation of benzene-1,4-diol (hydroquinone) [24]. Fig. 4 shows the rates of the oxidation reaction of benzene-1, 4-diol, the intensity of PL sub-band at 580 nm (2.1 eV) attributed to the surface hydroxyl group and content of CdS nanoparticles incorporated into TiO$_2$ matrix. It can be seen that the highest photoactivity is exhibited by composites with the lowest visible PL emission and the lowest CdS nanoparticle content. For a composite with short aging time and increased CdS content, a delayed decrease in photoactivity is observed, seemingly due to the competition of several alternative effects – (I) an increase in visible PL emission associated with harmful radiative recombination at the surface defect states, (II) an increase in the number of TiO$_2$/CdS heterojunctions which can improve the transfer and separation of charge carriers and (III) an increase in the number of titania surface defects serving as centers of photo-induced redox reactions.

4. Conclusions

The paper explores the effect of gel aging time on visible light photoluminescence of TiO$_2$/CdS composites fabricated by modified sol-gel route using pre-synthesized CdS colloidal nanoparticles. For this, a series of TiO$_2$/CdS and bare TiO$_2$ powders prepared with varying gel aging time as well as bare CdS coagulated nanoparticles were studied. TiO$_2$/CdS and TiO$_2$ exhibit an amorphous-nanocrystalline structure, and with an increase in gel aging time, the content of amorphous titania decreases as phase transformation from amorphous into crystalline titania proceeds and the content of incorporated CdS nanoparticles decreases.
It was found that for sol-gel TiO$_2$/CdS powders, gel aging time at boiling point is a crucial parameter to affect visible light PL emission. With increasing aging time, visible PL emission associated with TiO$_2$ surface defects such as oxygen vacancy and hydroxyl group as well as yellow-green and red emission is suppressed, related to the defect states of CdS nanoparticles. At short aging times, a significant non-additive increase in the intensity of visible luminescence bands of both phases is observed, which is considered as a result of the increased structural imperfection in the TiO$_2$/CdS interface, especially at the early stages of aging. Surface defect states and concentration of CdS nanoparticles are concluded to significantly affect visible PL emission, unlike the crystallinity degree of titania.

There is a correlation between visible PL emission and visible-light photoactivity of TiO$_2$/CdS composites. Composites with suppressed visible PL emission exhibit the enhanced visible-light photoactivity.

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