1. Introduction

Understanding the early stage in the precipitation pathway is of fundamental relevance in order to achieve an appropriate control at the macroscopic level on the solid formation, for instance in terms of morphology, physicochemical properties, and crystalline phase.\cite{1,2} CaCO$_3$ is a model system, archetype of several sparsely soluble inorganic materials. The precipitation pathway includes the nucleation and growth of a solid phase from supersaturated solutions. At the early stage, even the physical nature of the entities with a density higher than that of the solvent is a matter of debate.\cite{7}

Precipitation is a complex process. Homogeneous primary nucleation is the first elementary process to consider in solid formation pathway.\cite{8} There are two descriptions developed for such phenomenon occurring in solutions: classical nucleation theory (CNT) and nonclassical nucleation theory (NCNT).\cite{5,9,10} These theories have been recently reviewed.\cite{12}

In both cases, the formation of embryos or clusters starts with a reacting solution of dissolved aqueous species, triggered by a change of the system's saturation. Afterward, these entities grow and the reaction may follow several pathways over a period of time, according to the Ostwald's rule of stages, until the final thermodynamically stable product is achieved.

Calcium carbonate is a model system to investigate the mechanism of solid formation by precipitation from solutions, and it is often considered in the debated classical and nonclassical nucleation mechanism. Despite the great scientific relevance of calcium carbonate in different scientific areas, little is known about the early stage of its formation. Therefore, contactless devices are designed that are capable of providing informative investigations on the early stages of the precipitation pathway of calcium carbonate in supersaturated solutions using classical scattering methods such as wide-angle X-ray scattering (WAXS) and small-angle X-ray scattering (SAXS) techniques. In particular, SAXS is exploited for investigating the size of entities formed from supersaturated solutions before the critical conditions for amorphous calcium carbonate (ACC) nucleation are attained. The saturation level is controlled and kept constant by mixing four diluted solutions (i.e., NaOH, CaCl$_2$, NaHCO$_3$, H$_2$O) at constant $T$ and pH. The scattering data are collected on a liquid jet generated about 75 s after the mixing point. The data are modeled using parametric statistical models providing insight about the size distribution of denser matter in the liquid jet. Theoretical implications on the early stage of solid formation pathway are inferred.
clusters) exist and they may aggregate or coagulate into larger entities. In the specific case of calcium carbonate, generally the first solid product has an amorphous nature (amorphous calcium carbonate, ACC), which eventually transforms toward more thermodynamically stable crystalline phases.[13–19] Some authors claim that such intermediate amorphous phase has some kind of local ordering, which depends on the pH value and defines the crystal phase that may form by its evolution, naming it as proto-calcite or proto-vaterite.[14] According to our recent findings,[20] pH has an influence on the calcium-to-carbonate ratio in the amorphous phase as well as in the precursor entities, which are in local equilibrium equilibrium with the solution. This ratio might be the reason for the preferential phase obtained upon crystallization.

Indeed, the existence of subcritical embryos is expected even following the CNT reasoning with a statistical distribution of sizes, which is influenced by the saturation level. Therefore, other authors claim that there is no preferential size and thus the nucleation is classical.[21] The common point is that it is generally accepted that subcritical embryos or clusters exist. Using advanced analytical tools, such as cryoTEM or analytical ultracentrifugation (AUC), small entities were identified.[11,18] Nevertheless, sample manipulation is necessary and considering that such entities are in equilibrium with the liquid in which they are formed, any small physicochemical modification can strongly influence their stability. As a consequence, the investigations of these labile entities need to be done in situ, without modifying the equilibrium conditions that lead to their formation. The needed information is not merely the existence of subcritical entities, since this is considered a matter of fact, but their size distribution and nature (e.g., density) before the critical point, which can reflect the existence of a preferential size or a statistical distribution of size or even a different scenario.

Synchrotron-based scattering techniques play an important role in extracting information from entities suspended in the liquid phase. They are very powerful techniques thanks to the high brightness and flux and modern photon counting detectors. X-ray scattering is sensitive to the spatial variations of the sample electron density. In particular, SAXS is sensitive to density variations on the scale of nanometers to micrometers, providing very strong, meaningful, and informative scattering information at small angles,[22] yielding information on particle size and size distribution.[23,24] At wide-angle X-ray scattering (WAXS), the scattering yields information on a subnanometric scale, such as the atomic structure, more detailed information on crystallinity and defects, and elaborate size and shape distribution information. A limiting factor of WAXS is that typically, the diffraction signal is weaker by several orders of magnitude when compared to SAXS, the latter remaining as the only method exploitable when the sample is characterized by a combination of weak contrast and tiny concentrations, as is the case for near-saturation CaCO₃ solutions. The scattering process follows the reciprocity law: scattering information from larger features corresponds to lower angles.[25,26] Therefore, the hardware used for such measurements needs to be carefully designed in order to define the optimal trade-off among energy and flux of the beam, geometry, and detector resolution.

Ballauff and co-workers demonstrated the effectiveness of SAXS specifically on the investigation of ACC, applying a stopped-flow device.[23,24] With that setup, two solutions of calcium carbonate ions, both in the concentration range between 7 and 9 × 10⁻³ m, were rapidly mixed in equal volumetric proportion and transferred into a quartz capillary. The formation of ACC—and its transformation into crystalline phases—was investigated over time at pH = 10.5 and T = 25 °C. Analyzing their experimental conditions, considering the most recent thermodynamic database[27] and the value for ACC solubility in the described condition (≈1.80 × 10⁻⁴),[20] the experimental saturation levels corresponds to values of 57 and 74, respectively (Supporting Information). It is worth noting that the saturation here is calculated as S = IAP/Ksp where IAP stands for ionic activity product and Ksp is the solubility product of ACC. The same experimental conditions were used in the work of Huber and co-workers, where the ACC formation was investigated in a cuvette.[28]

At these very high saturation levels, the spinodal decomposition may occur, the ACC formation rate is very high, the solution mixing plays a fundamental role on the obtained particle size distribution, the ACC transforms to crystalline phases in a time scale of tens to hundreds of seconds, and the heterogeneous nucleation on the reactor wall is highly probable, be it a capillary or a cuvette. Consistently, experimental evidences associated with the aforementioned processes were addressed by the authors.[21,24,28,29] Such experimental conditions can be effectively utilized when the aim of the study is focused on the ACC particles, their size, and nature, as well as their dynamic transformation into crystalline phases. On the contrary, in the present work, we focus our study on a different timeframe, i.e., the one corresponding to the stage before the critical condition for the formation of ACC particles by homogeneous nucleation. Therefore, the saturation level needs to be much closer to unity and the system has to be studied in steady state (and not dynamic) mode. The investigation of the system at saturation levels down to values <4 (or even <1, i.e., undersaturated) poses experimental challenges. First, the concentration of the entities under investigation strongly decreases (up to two orders of magnitude lower compared to the work of Ballauff and co-workers). In order to achieve an appropriate signal-to-noise ratio, long data acquisition time (several minutes) at constant conditions of S, pH, and T need to be accomplished. Therefore, any stopped-flow techniques is precluded (being designed to investigate dynamic processes). Second, due to the labile signal of the target amorphous entities, the background generated by the sample needs to be minimized, the influence of the beam on the sample excluded (beam damage), and heterogeneous nucleation on the walls of the capillary (or any other cell) ruled out.

To face this experimental challenge, a pulsation-free micrometer-sized reactive horizontal liquid jet setup was built. The liquid jet is not confined by a solid, making it a contactless setup not susceptible to the wall effects.[30–33] Additionally, the liquid interrogated by the beam is continuously refreshed so the beam damage is intrinsically excluded and the dynamic mixing conditions allow a time-independent setting of concentrations of chemicals, T, pH, S, and ionic strength. Besides, the setup was calibrated for absolute scale with a standardized nanoparticle suspension.[34]
An accurate chemical speciation model is mandatory for a precise calculation of the system saturation level. We applied the developed thermodynamic model for ACC precipitation, which can solve the speciation as well as compute the solubility of ACC, in the specific experimental conditions.\cite{38} Even if the experimental conditions were set before the critical point for ACC formation, we refer the ACC as reference material to calculate the saturation level. Upon saturation increase above the critical conditions, the system may evolve toward the formation of ACC, establishing a link with the studies carried out by Balauff and co-workers and Huber and co-workers\cite{21,24,28,29} at higher saturation and our previous investigation.\cite{20}

In this paper we present both the results on cluster size distribution as a function of the saturation level with respect to ACC and the appropriate setup to carry out such in situ measurements also on other systems.

2. Data Analysis

Before their analysis, the experimental data need a careful conversion, from raw data (photon counting) into diffraction data (intensity-angle; intensity-momentum transfer Q on regular step). Scattering patterns were collected with reasonable acquisition times at the selected saturation levels, as well as from pure water, and from air. The system was washed with diluted acetic acid, followed by water, before each acquisition. Ancillary data (S, time-delay between the mixing point, and measuring point, T, pH, and jet size) were also recorded. In following sections, the applied calibration on the SAXS data and fitting approaches are described. As usual, the scattering data were collected in arbitrary units. A suitable standard was measured in order to obtain an appropriate scale calibration factor and convert the data to absolute units. This standard was a spherical gold nanoparticle (AuNP) suspension of known concentration (200 mg L^{-1} as Au) and narrow size distribution (18 nm, evaluated by TEM),\cite{35} measured using the same liquid jet and instrument setup as for the samples under investigations. The scale calibration factor was evaluated to \(1.011 \times 10^{-17}\).\cite{36}

2.1. Fitting Approaches

The entities in suspensions were assumed to consist of polydisperse particles with a spherical shape.\cite{37} A parametric distribution model was used for modeling and analyzing the collected data.\cite{38}

Two approaches were attempted. First, a more classical approach was evaluated. We assumed a log-normal size distribution of spherical clusters of unknown density. The scattering intensity of this polydisperse system is weighted by a two-parameter log-normal distribution. The observed data (with a measured background pattern whose relative scale was set free) for the saturation levels \((C_0, C_s)\) were fitted very closely with the model intensity, but the results presented several weak points (extremely broad distributions, little correlation between samples, and not all the fits converged). This approach was rejected in favor of a slightly more conceptually complex one (bimodal approach), albeit without increasing the number of parameters.

2.2. Bimodal Approach

In this approach, we assumed that the whole system is composed of two populations of particles: one population includes monodisperse small dense clusters (SDC hereafter), with diameter of about 2 nm; a second population includes polydisperse aggregates (i.e., superclusters) with higher volume, formed by several SDC loosely packed (without coalescence). Reasonable values of packing fraction range between 50% and 72%. The lower limit still gives some compactness; the upper limit is the close-packing limit, which is still below the coalescence region. We assume a packing fraction of 70% (vol.) of SDC in the aggregates, near the close-packing limit, and a minimum diameter of \(\approx 5\) nm (i.e., a supercluster contains at least 8 SDC).

The contributions of these two populations lead to two linear components for the intensity (one from the population of the SDC and one from the population of the superclusters), plus the separately measured background \((C_0)\).

The contribution from the SDCs, as they result to be rather small, is a relatively flat and featureless trace in the experimental \(Q\) range (0.015–0.07 Å^{-1}) and it is weak and highly correlated with the background; therefore, i) the scale of this trace cannot be precisely determined and ii) the SDC diameter—approximated as monodisperse spheres—can be determined within a 20–30% error, which may also be intrinsic in the SDC nature. Nevertheless, the fact that this signal is refined to consistent values (diameter of about 2 nm, corresponding broadly to the monomers detected by Huber and co-workers) in all of our best experimental patterns (8 series of data) for all experimental conditions, adds credibility to their existence, at least qualitatively.

For the supercluster (polydisperse aggregates), the size distribution function has been refined both as a lognormal and an exponential distribution (which is a limiting case of the Schulz–Flory a.k.a. Gamma distribution\cite{39–41})

\[
P_m = C \exp(-D_m/D_0) \tag{1}
\]

the latter distribution being then selected as giving the most credible results. Here \(C\) is a normalization constant such that the number fractions \(P_m\) sum to 1, \(D_0\) is the distribution parameter, \(D_m = m\delta\) is the diameter of the \(m\)th cluster (variable between 5 and 100 nm), and \(\delta\) is a convenient step representing the diameter of a sphere containing one SDC and an equal volume of water. Here, the role of \(D_m\) is more a computational detail than a meaningful physico-chemical detail. Hereby, we are not making a strict structural hypothesis on the superclusters. In fact, when dealing with continuous size distribution of particles, one must evaluate the distribution at discrete values of the diameter \(D\), which are finely enough spaced to give meaningful results but not so much as to overburden the calculations. In our case of larger spheres (i.e., spheres built as loose coalescence of much smaller predefined spheres), such diameter steps are chosen so that in the \(m\)th spherical shell of thickness \(\delta/2\) there are \(3m^2 - 3m + 1\) smaller particles, and accordingly, in the \(m\)th sphere of diameter \(D_m = m\delta\) there are \(m^3\) smaller particles. This treatment represents a robust and convenient way of defining the sampling step for size distributions.\cite{42} Accordingly, to evaluate \(\delta\),
we use a value (70%) of the packing density that is within the most meaningful region (30% to 72% for loose to close packing without coalescence).

The scattering intensities must be summed over all the particles sizes and weighted by their size distribution function $P_m$. Hence, the calculated scattering intensity of this polydisperse system is given by

$$I(Q) = k \sum_m P_m V_m^2 \left(3 \frac{\sin(QR_m) - QR_m \cos(QR_m)}{(QR_m)^3}\right)^2$$

where $m$ is indexing over the possible different spherical clusters (see below), $N$ is the total number of particles in the beam, $R_m = D_m/2$ is the sphere radius, $Q$ is the transferred momentum, $V_m$ is the volume of the clusters, and $\Delta \rho$ is the clusters electron density contrast. At $Q = 0$, the scattering forward intensity is given as

$$I(Q = 0) = k \sum_m P_m V_m^2$$

with $k$ representing the calculated scale factor to bring absolute intensities onto the experiment scale, estimated by a reference material as mentioned above. In order to fit the data with the model, the free parameters were the exponential size distribution ($D_0$), the scale factor of the large aggregates distribution ($S_{agg}$), the scale of the background trace to be subtracted ($S_{bkg}$), and the scale of an additional diffraction trace of a fixed small-diameter sphere representing the SDC ($S_{SDC}$). The scales $S_{agg}$ and $S_{SDC}$ always resulted highly correlated (within $10^{-4}$ from perfect correlation), so their separate values have no meaning. As such, the scale $S_{SDC}$ does not allow quantification of free-standing SDC. However, eliminating their contribution from the fit has negative effect on the stability of results; therefore, they were kept in, at least as an additional background term. It is likely that their contribution is present, but they cannot be quantified. The scale of the measured background always resulted very close to 1 as expected. Only small and unaccounted for parasitic effects (e.g., tiny jet diameter variations, monitor error) could affect it. A simple grid search algorithm could evaluate: the concentration of Ca ions ($C_{Ca^2+}$) in superclusters to be compared with the concentration of CaCO$_3$ ion pair present in solution ($C_{CaCO3}$); the average diameter weighted by the mass distribution ($D_{MD}$); the average width of the mass distribution (r.m.s. width-MD); the diameter at maximum of the mass distribution ($D_{MD}$); the average diameter weighted by the number distribution ($D_{ND}$);

The reliability and significance of the experimental SAXS data were checked for all the measurements conducted at the different saturation conditions using the two approaches discussed above. The subject of our study is low saturation ($S < 4$) calcium carbonate solutions. In our experimental condition, the IAP is $<8 \times 10^{-8}$ and the estimated concentration of CaCO$_3$ ion pair is $<8$ ppm (Table 1). Therefore, assuming that all ion pairs form SDC and considering the lower scattering contrast of entities that might be formed in these conditions, a meaningful signal must be 2–3 orders of magnitude weaker relative to the liquid carrier signal. With respect to Ballauff and co-workers and Huber and co-workers, our concentration range is set more diluted up to two orders of magnitude and the cluster electron density contrast was also about 2–3 times smaller. While we still obtain a useful signal, with robust signal-to-noise ratio in the Guinier region and in the Fourier region, the Porod region signal is too low to apply the Porod invariant method.

We also attempted a quantification of Ca$^{2+}$ contained in the superclusters. The SDC were ignored, as not quantifiable. For the superclusters, the scale and the exponential diameter distribution parameter could be properly refined. As a summary, we could evaluate:

3. Results and Discussion

As indicated in Equation (3), the scattering intensity is proportional to the square of the scattering contrast, which represents the excess electron density of the suspended particles with respect to the surrounding medium. SDCs are poly-hydrated CaCO$_3$ ion pair; the composition is generally CaCO$_3 \cdot n$H$_2$O with atomic weight of 100.1 + 18.02$n$ and 50 + $10n$ electrons per unit formula.

Water was found to be a key factor for ACC nucleation. This raised a suggestion of using a different water fraction for every saturation level. The water content and the cluster density at each condition ($C_n$) were estimated applying the thermodynamic model, which take into account for different water content. We fixed 22 CaCO$_3$ formula unit per cluster and a number of water molecules per Ca atom of 18, 12, and 8 for conditions $C_2$, $C_4$, and $C_8$, respectively. The calculated SDC mass densities for conditions $C_2$, $C_4$, and $C_8$ are estimated to be 1.160, 1.227, and 1.314 g cm$^{-3}$, corresponding to SDC sizes as 2.94, 2.62, and 2.35 nm, respectively (Supporting Information). Therefore, SDCs have a distinct $D_0$ at each condition, and the contrast of the respective superclusters is $\eta \Delta \rho$ where $\eta$ is the packing fraction (which we arbitrarily fixed to 0.7—near the close-packing fraction). Note that also the packing fraction could also very well be variable with the saturation. However, this at the moment must remain speculative.

Although our data are noisier than usual for a SAXS experiment (because of the very low concentrations and contrast), they showed to be informative and selective enough to clearly show a preference for an exponential distribution of superclusters with respect to a single-size and even with respect to a classical log-normal. The exponential distribution fits the result quite nicely from any perspective, meaningfulness of results, graphical appearance, and especially the statistical indicators. In this respect, including the contribution due to SDC as aforementioned discussed, the goodness-of-fit, GoF, is close to 1 (the ideal value, see Table 1) in all cases; moreover, practically no point deviates more than 3$\sigma$ from the refined model value, as shown in Figure 1.

3. Results and Discussion

Part. Part. Syst. Charact. 2019, 36, 1800482
the average width of the number distribution (r.m.s. width-ND);
- the exponential distribution parameter ($D_0$);
- the scale factors of the SDCs ($S_{SDC}$), superclusters ($S_{agr}$) and background ($S_{bkg}$).

Moreover, the following correlations were estimated:
- the background with superclusters ($I_{bkg-aggr}$);
- the background with SDC ($I_{bkg-SDC}$);
- the aggregates with SDC ($I_{agr-SDC}$); and
- $I_{agr}(Q = 0)$, $I_{SDC}(Q = 0)$, and the product $Q_{min} \times R_g$.

Most of these values are reported in Table 2 for the selected conditions C2, C4, and C8.

As discussed, the experimental evidences reveal that the SDC signal is poorly defined, being it a rather flat trace highly correlated with the background. Therefore, size and quantity of SDC are affected by very high uncertainty to be considered meaningful. On the other hand, we can quantify Ca ions from the superclusters because the signal is much more structured (i.e., the large entities can be easily and precisely quantified). Consequently, the Ca ions concentration ($C_{Ca}$, Table 2)—omitting the unknowable fraction of the SDC—is below the calculated value ($C_3$), very near and slightly above it ($C_4$) and very underestimated ($C_8$, probably because of the formation of large entities). In this sense, we assumed a unimodal population of small ($\approx 2$ nm) (with density 1.16 to 1.31 g cm$^{-3}$, Supporting Information) spherical clusters and a population of less dense but larger superclusters (broad size distribution up to 100–200 nm) constituted by the SDC with $\approx 30\%$ water existing in between them (Figure 2). A reasonable packing fraction limit of about 70$\%$ was assumed, although a larger and possibly variable packing fraction would be more correlated.
aggregates contributions are given in lieu of the raw scale factors. The intensities extrapolated at $Q = 0$ —the distribution parameter—as it is useful to evaluate the material under investigation, we can state that the experimental results are consistent with the presence of clusters characterized by a diameter of about 2 nm that might be in presence of a population of SDC seems consistent with other experimental results\textsuperscript{[11,18,47]} and compatible with the calculated critical size.\textsuperscript{[20]} On the other hand, the presence of highly hydrated large superclusters seems consistent with a liquid-like separation model.\textsuperscript{[44]} Therefore, within the approximation and assumption included in the data analysis, combined with the intrinsic limitation of technique and the properties of the material under investigation, we can state that the experimental results are consistent with the presence of clusters characterized by a diameter of about 2 nm that might be in a high numerical concentration but almost negligible overall mass, and a population of superclusters, limited in number concentration but containing the majority of the $\text{CaCO}_3^0$ ion pair in solution.

For a so described system, at the early stage of the denser matter formation, it is not a surprise that by cryoTEM the limited number of large superclusters might be excluded by sampling and they are not detectable by AUC. On the other hand, the computational model might emphasize the presence of superclusters (or nanodroplets of denser liquid phase), corresponding to the form in which the majority of $\text{CaCO}_3^0$ ion pair mass is accumulated.

The beauty of the presented techniques and the data analysis approach is that the aqueous system under observation is not perturbed by any sampling or manipulation and the direct evaluation of superclusters at constant $S$, pH, and $T$ were achieved. Moreover, even if affected by large uncertainty, the presence of SDC is also consistent with the experimental results. The same approach can be extended and applied to other systems, paving the way toward a more comprehensive understanding of the early stage of solid formation and merging only apparently divergent conclusions based on other investigation (and complementary) techniques.

### Table 1. Calculated saturation levels with respect to ACC solubility ($K_{\text{sp}} = 2.58 \times 10^{-8}$) for the four experimental conditions considered.\textsuperscript{[20]}

| Parameter | $C_0$ | $C_1$ | $C_2$ | $C_3$ | $C_4$ |
|-----------|-------|-------|-------|-------|-------|
| $C_{\text{CaCl}_2}$ flow rate [mL min$^{-1}$] | 0.00 | 0.40 | 0.80 | 1.61 |
| $C_{\text{NaHCO}_3}$ flow rate [mL min$^{-1}$] | 3.72 | 3.67 | 3.62 | 3.52 |
| $C_{\text{Water}}$ flow rate [mL min$^{-1}$] | 0.90 | 0.92 | 0.95 | 1.00 |
| $C_{\text{NaOH}}$ flow rate [mL min$^{-1}$] | 3.38 | 3.00 | 2.63 | 1.88 |
| $C_{\text{CaCl}_2}$ flow rate [mL min$^{-1}$] | 0.00 | 1.00 $\times 10^{-4}$ | 2.01 $\times 10^{-4}$ | 4.02 $\times 10^{-4}$ |
| $C_{\text{NaHCO}_3}$ flow rate [mL min$^{-1}$] | 9.31 $\times 10^{-4}$ | 7.35 $\times 10^{-2}$ | 7.24 $\times 10^{-2}$ | 7.04 $\times 10^{-2}$ |
| $C_{\text{Water}}$ flow rate [mL min$^{-1}$] | 0.90 | 1.09 $\times 10^{-2}$ | 2.22 $\times 10^{-2}$ | 4.56 $\times 10^{-2}$ |
| $C_{\text{NaOH}}$ flow rate [mL min$^{-1}$] | 3.38 | 7.97 $\times 10^{-5}$ | 1.61 $\times 10^{-4}$ | 3.26 $\times 10^{-4}$ |
| $C_{\text{CaCl}_2}$ ion [mL$^{-1}$] | 5.60 $\times 10^{-4}$ | 5.60 $\times 10^{-4}$ | 5.60 $\times 10^{-4}$ | 5.59 $\times 10^{-4}$ |
| $C_{\text{NaHCO}_3}$ ion activity [mL$^{-1}$] | 6.65 $\times 10^{-1}$ | 6.63 $\times 10^{-1}$ | 6.60 $\times 10^{-1}$ | 6.48 $\times 10^{-1}$ |
| $C_{\text{Water}}$ ion activity [mL$^{-1}$] | 6.54 $\times 10^{-3}$ | 6.53 $\times 10^{-1}$ | 6.51 $\times 10^{-1}$ | 6.48 $\times 10^{-1}$ |
| $C_{\text{NaOH}}$ ion activity [mL$^{-1}$] | 3.66 $\times 10^{-4}$ | 3.66 $\times 10^{-4}$ | 3.65 $\times 10^{-4}$ | 3.62 $\times 10^{-4}$ |
| $C_{\text{CaCO}_3}$ ion pair [ppm] | 0.00 | 2.04 | 3.96 | 2.04 |
| $C_{\text{Water}}$ ion activity [mL$^{-1}$] | 2.04 | 3.96 | 3.62 $\times 10^{-4}$ | 7.44 |
| $C_{\text{NaOH}}$ ion activity [mL$^{-1}$] | 1.94 $\times 10^{-8}$ | 3.88 $\times 10^{-8}$ | 7.80 $\times 10^{-8}$ | 1.51 |
| Saturation [$I$] | 0.00 | 0.75 | 1.51 | 3.02 |
In the specific case of calcium carbonate formation pathway, before ACC nucleation, we speculate an increase of supercluster density with the saturation level and, therefore, the involvement of water in the formation pathway. In the literature, Raiteri et al. have already reported the influence of water as key factor for the nonclassical nucleation of ACC.[43] The authors concluded that their calculations are consistent with the formation of a narrow distribution of small clusters during the early stage of the process, that then agglomerate to larger entities. A Gibbs free energy landscape giving more than one point with zero derivative, so implying a nonclassical nucleation pathway, is also proposed.

Our experimental data are consistent with the presence of small and large clusters, but the representation of the process is formally different. Figure 3 summarizes the overall concept for the ACC formation pathway from the point of view of the solution chemistry and population balance approach.[20] The black line represents the measured free Ca$^{2+}$ ion in solution, the violet line is the overall amount of CaCO$_3^-$ ion pair (as isolated entity and clusters) and the blue line reports on the amount of ACC formed. The red circles represent the experimental conditions discussed in this paper, all belong to the stage I of the solid formation process, i.e., under (pseudo)thermodynamic equilibrium and before ACC nucleation. The dashed line denotes the solubility limit of ACC, experimentally evaluated from the plateau of the free Ca$^{2+}$ curve for time $>$120 min. Indeed, C$_2$ corresponds to undersaturated conditions (below the dashed line) whereas C$_4$ and C$_8$ correspond to supersaturated conditions, but with $S$ lower than the critical point for ACC nucleation (maximum of the black curve). In the figure, the reported squared areas depict the populations of SDCs and superclusters in solution at each stage. Even when the system is undersaturated, both SDCs and superclusters exist. The system is highly hydrated, the SDC themselves are denser than water and aggregated in hydrated superclusters with a broad size distribution. As the system saturation increase, both the...
density of the SDC and the size of the superclusters increase. This process gradually proceeds until the critical conditions for primary nucleation, which corresponds to a critical density of the SDC within the superclusters. In the figure, the yellow particle denotes the primary nucleation event, i.e., the transformation of a SDC to an ACC primary nucleus by dehydration. Almost simultaneously, secondary nucleation occurs within the superclusters (red spheres), leading to the massive ACC formation. The process after the maximum of the black curve is described in details in ref. [20]. Here, we complete that investigation with the conditions C2, C4, and C8.

Even if we are not able to figure out the details of the Gibbs free energy landscape, it seems clear that when the saturation of a system is gradually raised, the system reacts defining new (quasi)equilibrium states at each S levels, where a population of entities with a large variety of size and density is present. With this picture in mind, the representation of the Gibbs free energy landscape, as a well-predefined and fixed profile that is carefully followed during the entire precipitation pathway, results obsolete. In fact, to each S value, a new function for both the volume and the surface contributions to the overall Gibbs free energy—and therefore new size distribution of denser entities—corresponds. Thus, the appropriate Gibbs free energy landscape cannot be plotted in a 2D plane, since it is function of size and saturation, outlining a more complex—at least 3D—representation for the solid formation pathway.

It is worth mentioning that if the saturation is further increased above the critical point and the increase is slow enough, the system might evolve according to schematization of Figure 3. The equations associated to the classical nucleation theory can still be used to estimate the nucleation rate since they are unaffected by the details of undercritical events.

4. Conclusion

SAXS technique combined with a liquid jet setup, an accurate thermodynamic model, and a bimodal particle population approach have allowed the detection of superclusters in situ. As a model system, the calcium carbonate formation pathway was investigated at different saturation levels, including undersaturated conditions.

The experimental results are consistent with the presence of a broad size distribution of entities (superclusters) with variable density and highly hydrated, which were postulated by the computational model. Simultaneously, a population of clusters of about 2 nm were qualitatively detected, previously identified in the literature using cryoTEM and AUC, which are mainly aggregate in the superclusters. The presented results are not only consistent with the literature, but for the first time, both populations of denser matters in solution are simultaneously detected by SAXS, before the critical condition for ACC precipitation. Such entities are present even in undersaturated conditions (with respect to ACC). The populations of SDC and supercluster evolve increasing the saturation level toward higher density (dehydration) and larger superclusters.
Because size and composition of superclusters change with the saturation, the conceptual representation of the Gibbs free energy landscape should be revisited according to a surface in a 3D space. Therefore, the concept of CNT and NCNT should be entirely reconsidered.

5. Experimental Section

Calcium chloride, sodium hydroxide, and sodium bicarbonate were purchased from Sigma Aldrich (analytical grade, ReagentPlus). Aqueous solutions CaCl₂ (2 × 10⁻³ mol L⁻¹), NaOH (5 × 10⁻³ mol L⁻¹), NaHCO₃ (20 × 10⁻³ mol L⁻¹) and pure water were prepared using CO₂-free milliQ water. A pulsation-free micrometer-sized horizontal reactive liquid jet setup was specifically built for the measurements. The system as a whole was composed of four HPLC pumps—each of them equipped with a pulsation damper system and high precision Coriolis mass flowmeters, a mixing system, a delay loop, and a catcher (Figure 4). A micromixer manifold equipped with five inputs and one output was used to mix the solutions. The fifth input was connected to an additional HPLC pump delivering a 10 wt% acetic acid solution, which serves to clean the system. The manifold exit held the delay loop, which consisted of a Teflon tube of a certain length and internal diameter. The loop defined the delay time between the mixing point and the irradiated liquid jet. Delivering tubes before the mixer and the delay loop were thermostated using a double-walled water-jacketed tubing system. The delay loop outflow was connected to a second manifold and a capillary (nozzle), which can be chosen between different materials and internal diameters. The delay time was fixed to 75 s. The internal capillary diameter defined the overall flow rate for a defined pressure drop. In this study, stainless steel capillaries with 250 µm internal diameter and an overall flow rate of 8 mL min⁻¹ were used. The pumping system was remotely controlled and monitored. A catcher, collecting the ejected liquid after X-ray exposure, was equipped with a micro stirrer, a pH electrode, and a Pt1000 sensor.

Experiments were carried out at different flow rates for each delivery line in order to achieve the specific pH and S values. The flow rates were preliminarily calculated based on the thermodynamic model and verified through the experimental pH values within an accuracy of ±0.01 pH unit. The speciation model considers the formation of CaCO₃ ion pairs, which may form clusters, and it excludes the formation of the ACC phase.[20] The agreement between the predicted and the experimental pH values confirmed that ACC precipitation does not occur. In fact, in the case of ACC formation, the pH decreased rapidly and a higher amount of NaOH was needed to keep the pH at the set point. Consistently, ACC precipitation occurred if S was increased above the critical point, readily recognized by the pH probe in the catcher. These experimental evidences confirmed the validity of the thermodynamic model and excluded the formation of ACC in the experimental conditions.
Hence to guarantee that the S values were correctly set, an accurate thermodynamic model of the system, an accurate calibration of the pH meter (calibrated close to the pH of interest and at the same T and ionic strength of the experiment), and the precise concentrations of chemicals were needed. Moreover, the precise on-line monitoring of pH and T and a stable pulsation-free flow rate control of the four chemicals were fundamental to keep the saturation of the system constant during the entire measurement timeframe.

Table 1 summarizes the experimental conditions and the resulting saturation levels. Four experimental conditions were analyzed in detail hereafter named as $C_0$, $C_2$, $C_4$, and $C_6$ for $0 \leq S \leq 3.02$. These values defined the experimental conditions before the critical point for the homogenous nucleation of ACC, which corresponded to $S = 3.7$ at pH 9.0 and T = 25.0 °C (Figure 3D, in ref. [20]). In particular, $C_2$ corresponded to undersaturated conditions and $C_0$ represented the blank, periodically measured as background signal.

Synchrotron small-angle X-ray scattering measurements were carried out at the Material Science beamline (X04S-MS) of the Swiss Light Source (SLS) at PSI. This synchrotron station was built for WAXS powder diffraction measurements but it also had some SAXS capabilities. Modern photon counting detectors (Mythen II & PILATUS 6M) were installed. The liquid jet, horizontal and orthogonal to the X-ray beam, was mounted on a double micrometric translation stage and optically installed. The liquid jet, horizontal and orthogonal to the X-ray beam, was set at 9.5 KeV (λ = 0.13 Å) where the X-ray flux was maximal. Data were collected with the Mythen II detector system [19] with its 0.0036° pixel, it had a sufficient resolution for SAXS on this system with a minimum accessible 2θ scattering angle of ~0.18–0.20°, corresponding to a minimum accessible momentum transfer of $Q = 4 \pi \sin(\theta)/\lambda = 0.016 \text{ Å}^{-1}$ where $\lambda$ is the wavelength of incident X-ray.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

A.S.A.M. and A.C. have contributed equally. The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Conflict of Interest

The authors declare no conflict of interest.

Keywords

calcium carbonate, CNT, contactless devices, NCNT, precipitation pathway, SAXS

Received: November 9, 2018
Revised: March 27, 2019
Published online: May 23, 2019

[1] T. H. Zhang, X. Y. Liu, Angew. Chem., Int. Ed. 2009, 48, 1308.
[2] C. Ratsch, J. A. Venables, J. Vac. Sci. Technol., A 2003, 21, 596.
[3] P. G. Vekilov, Prog. Cryst. Growth Charact. Mater. 2016, 62, 136.
[4] Y. K. La Mer, Ind. Eng. Chem. 1952, 44, 1270.
[5] H. Cölfen, M. Antonietti, Mesocrystals and Nonclassical Crystallization, John Wiley & Sons Ltd, Chichester, West Sussex, UK 2008.
[6] V. J. Anderson, H. N. W. Lekkerkerker, Nature 2002, 416, 811.
[7] Q. Hu, M. H. Nielsen, C. L. Freeman, L. M. Hamm, J. Tao, J. R. I. Lee, T. Y. J. Han, U. Becker, J. H. Harding, P. M. Dove, J. J. De Yoreo, Faraday Discuss. 2012, 159, 509.
[8] E. J. W. P. Schmelzer, Nucleation Theory and Applications, J. Wiley & Sons, New York 2006.
[9] J. Frenkel, Kinetic Theory of Liquids, Clarendon Press, Oxford 1946.
[10] D. Gebauer, H. Cölfen, Nano Today 2011, 6, 564.
[11] D. Gebauer, A. Volkel, H. Cölfen, Science 2008, 322, 1819.
[12] S. Kartikha, T. K. Radhakrishnan, P. Kalaiavel, Cryst. Growth Des. 2016, 16, 6663.
[13] N. Abdel-Aal, K. Satoh, K. Sawada, J. Cryst. Growth 2002, 245, 87.
[14] W. N. Al Nasser, A. Shaikh, C. Morris, M. J. Houmslow, J. A. D. Salmon, Chem. Eng. Sc. 2008, 63, 1381.
[15] R. Demichelis, P. Räteri, J. D. Gale, D. Quigley, D. Gebauer, Nat. Commun. 2011, 2, 590.
[16] M. Donnet, A. Almara, J. Lemaître, P. Bowen, J. Phys. Chem. B 2010, 114, 12058.
[17] M. H. Nielsen, S. Aloni, J. J. De Yoreo, Science 2014, 345, 1158.
[18] E. M. Pouget, P. H. Bomans, J. A. Goos, P. M. Frederik, G. de With, N. A. Sommerrdijk, Science 2009, 323, 1455.
[19] B. B. Schroeder, D. D. Harris, S. T. Smith, D. O. Lignell, Cryst. Growth Des. 2014, 14, 1756.
[20] A. Carino, A. Testino, M. R. Andalibi, F. Pilger, P. Bowen, C. Ludwig, Cryst. Growth Des. 2017, 17, 2006.
[21] K. Henzler, E. O. Fetisov, M. Golib, M. D. Baer, B. A. Legg, C. Borca, J. M. Xto, S. Pin, J. L. Fulton, G. K. Schenter, N. Govind, J. I. Siepmann, C. J. Mundy, T. Huthwelker, J. J. De Yoreo, Sci. Adv. 2018, 4, eaao6283.
[22] T. Li, A. J. Senesi, B. Lee, Chem. Rev. 2016, 116, 11128.
[23] J. Bolze, B. Peng, N. Dingemouts, P. Panine, T. Narayanay, M. Ballauff, Langmuir 2002, 18, 8364.
[24] D. Pontoni, J. Bolze, N. Dingemouts, T. Narayanan, M. Ballauff, J. Phys. Chem. B 2003, 107, 5123.
[25] A. Guiner, G. Fournet, C. J. Walker, Small Angle Scattering of X-rays, J. Wiley & Sons, New York 1955.
[26] O. Glatter, O. Kratky, Small Angle X-ray Scattering, Academic Press, London 1982.
[27] OLI Studio, V9.2.4 2015, OLI System, Inc., Morris Plains, NJ, http://www.olisystem.com/.
[28] J. Liu, J. Rieger, K. Huber, Langmuir 2008, 24, 8262.
[29] J. Liu, S. Pancera, V. Boyko, A. Shukla, T. Narayana, K. Huber, Langmuir 2010, 26, 17405.
[30] J. Polte, T. T. Ahner, F. Delissen, S. Sokolov, F. Emmerling, A. F. Thümennach, R. Kraehnert, J. Am. Chem. Soc. 2010, 132, 1296.
[31] J. Polte, R. Erler, A. F. Thümennach, F. Emmerling, R. Kraehnert, Chem. Commun. 2010, 46, 9209.
[32] J. Polte, R. Erler, E. F. Thümennach, S. Sokolov, T. T. Ahner, K. Rademann, F. Emmerling, R. Kraehnert, ACS Nano 2010, 4, 1076.
[33] S. E. Wolf, J. Leiterer, M. Kappl, F. Emmerling, W. Tremel, J. Am. Chem. Soc. 2008, 130, 12342.
[34] S. E. Wolf, J. Leiterer, M. Kappl, F. Emmerling, W. Tremel, J. Am. Chem. Soc. 2016, 138, 1381.
[35] M. R. Andalibi, A. Testino, H. Hofmann, CHIMIA Int. J. Chem. 2016, 70, 457.
[36] A. S. A. Mohammed, A. Carino, A. Testino, M. R. Andalibi, A. Cervellino, J. Appl. Crystallogr. 2019, 52, 344.
[37] A. Carino, A. Walter, A. Testino, H. Hofmann, CHIMIA Int. J. Chem. 2006, 60, 47.
[38] A. Agbabiaka, M. Wiltfong, C. Park, J. Nanopart. 2013, 1, 1.
[39] Z. G. Fan, Y. G. Wu, X. H. Zhao, Y. Z. Lu, *Comput. Mater. Sci.* **2004**, 29, 301.
[40] S. Kumar, S. K. Kurtz, J. R. Banavar, M. G. Sharma, *J. Stat. Phys.* **1992**, 67, 523.
[41] M. Rogosic, H. J. Mencer, Z. Gomzi, *Eur. Polym. J.* **1996**, 32, 1337.
[42] A. Cervellino, C. Giannini, A. Guagliardi, *J. Appl. Crystallogr.* **2010**, 43, 1543.
[43] P. Raiteri, J. D. Gale, *J. Am. Chem. Soc.* **2010**, 132, 17623.
[44] A. F. Wallace, L. O. Hedges, A. Fernandez-Martinez, P. Raiteri, J. D. Gale, G. A. Waychunas, S. Whitelam, J. F. Banfield, J. J. De Yoreo, *Science* **2013**, 341, 885.
[45] L. Boldon, F. Laliberte, L. Liu, *Nano Rev.* **2015**, 6, 25661.
[46] G. Porod, *Kolloid-Z.* **1952**, 125, 51.
[47] P. J. M. Smeets, A. R. Finney, W. Habraken, F. Nudelman, H. Friedrich, J. Laven, J. J. De Yoreo, P. M. Rodger, N. Sommerdijk, *Proc. Natl. Acad. Sci. USA* **2017**, 114, E7882.
[48] P. R. Willmott, D. Meister, S. J. Leake, M. Lange, A. Bergamaschi, M. Boge, M. Calvi, C. Cancellieri, N. Casati, A. Cervellino, Q. Chen, C. David, U. Flechsig, F. Gozzo, B. Henrich, S. Jaggi-Spielmann, B. Jakob, I. Kalichava, P. Karvinen, J. Krempasky, A. Ludeke, R. Luscher, S. Maag, C. Quitmann, M. L. Reinle-Schmitt, T. Schmidt, B. Schmitt, A. Streun, I. Vartiainen, M. Vitins, X. Wang, R. Wullschleger, *J. Synchrotron Radiat.* **2013**, 20, 667.
[49] A. Bergamaschi, A. Cervellino, R. Dinapoli, F. Gozzo, B. Henrich, I. Johnson, P. Kraft, A. Mozzanica, B. Schmitt, X. T. Shi, *J. Synchrotron Radiat.* **2010**, 17, 653.