Voltage-Polarity Dependent Programming Behaviors of Amorphous In–Ga–Zn–O Thin-Film Transistor Memory with an Atomic-Layer-Deposited ZnO Charge Trapping Layer

Dan-Dan Liu, Wen-Jun Liu*, Jun-Xiang Pei, Lin-Yan Xie, Jingyong Huo, Xiaohan Wu and Shi-Jin Ding*

Abstract

Amorphous In–Ga–Zn–O (a-IGZO) thin-film transistor (TFT) memories are attracting many interests for future system-on-panel applications; however, they usually exhibit a poor erasing efficiency. In this article, we investigate voltage-polarity-dependent programming behaviors of an a-IGZO TFT memory with an atomic-layer-deposited ZnO charge trapping layer (CTL). The pristine devices demonstrate electrically programmable characteristics not only under positive gate biases but also under negative gate biases. In particular, the latter can generate a much higher programming efficiency than the former. Upon applying a gate bias pulse of +13 V/1 μs, the device shows a threshold voltage shift (ΔVth) of 2 V; and the ΔVth is as large as −6.5 V for a gate bias pulse of −13 V/1 μs. In the case of 12 V/1 ms programming (P) and −12 V/10 μs erasing (E), a memory window as large as 7.2 V can be achieved at 10⁵ of P/E cycles. By comparing the ZnO CTLs annealed in O₂ or N₂ with the as-deposited one, it is concluded that the oxygen vacancy (V̄O)-related defects dominate the bipolar programming characteristics of the TFT memory devices. For programming at positive gate voltage, electrons are injected from the IGZO channel into the ZnO layer and preferentially trapped at deep levels of singly ionized oxygen vacancy (V̄O⁺) and doubly ionized oxygen vacancy (V̄O²⁺). Regarding programming at negative gate voltage, electrons are de-trapped easily from neutral oxygen vacancies because of shallow donors and tunnel back to the channel. This thus leads to highly efficient erasing by the formation of additional ionized oxygen vacancies with positive charges.

Keywords: ZnO, In–Ga–Zn–O, Nonvolatile memory, Thin-film transistor (TFT), Oxygen vacancy

Background

A thin-film transistor (TFT) based on amorphous indium–gallium–zinc–oxide (a-IGZO) has been extensively studied for the application to flexible and transparent electronic systems [1–12]. This is attributed to some specific properties of a-IGZO films such as good uniformity, low processing temperature, visible light transparency, and high electron mobilities [13]. Other than that, a-IGZO TFT nonvolatile memories have also been proposed, and its nonvolatile data storage capability expands the scope of the a-IGZO TFT device utilization. As a typical architecture of nonvolatile memory devices, a floating-gated a-IGZO TFT memory has been intensively investigated in recent years. Up to now, various materials have been explored as a floating gate (i.e., charge storage medium), such as dielectrics [14, 15], metal nanocrystals [16, 17], and semiconducting materials [18–21]. Since a-IGZO is a natural n-type semiconductor, and hole inversion is hardly realized in an a-IGZO TFT under a negative gate bias, therefore, the a-IGZO TFT memories usually have a poor erasing efficiency. In other words, most a-IGZO TFT memories cannot be electrically erased through hole injection from the channel [14–16]. Nevertheless, Zhang et al. [21] fabricated a TFT memory using a-IGZO as both the charge trapping layer (CTL) and the channel layer, which exhibited electrically
programmable and erasable characteristics, as well as good data retention. Meanwhile, Yun et al. also investigated the characteristics of the a-IGZO TFT memories with different compositional IGZO CTL, revealing a decreasing memory window with increasing the O$_2$ partial pressure (P$_{O2}$) during sputtering deposition of the CTL [18]. In addition, Bak et al. reported the performance of the a-IGZO TFT memories with various conductivity ZnO CTLs and inferred that the optimized electronic nature of bandgap structure for the ZnO CTL could be one of the most important factors to realize highly functional oxide TFT memories [20]. Although the aforementioned oxide semiconductor CTL-based a-IGZO TFT memories exhibit superior electrical programming/erasing speeds, the bipolar programming characteristics of the abovementioned devices have not been reported, and the corresponding capture processes of different charges in the CTL of oxide semiconductor are not clear yet, especially for the trapping of positive charges.

In this work, a bipolar programmable a-IGZO TFT memory was fabricated by using an atomic-layer-deposited ZnO film as a CTL. By comparing the bipolar programming characteristics of the TFT memory devices with the as-deposited, O$_2$- or N$_2$-annealed ZnO CTLs, the capture processes of different charges in the ZnO layer were discussed. It is revealed that oxygen vacancy-related defects dominate the bipolar programming characteristics of the a-IGZO TFT memory devices.

Methods
P-type Si (100) wafers with resistivity of 0.001–0.005 Ω cm were cleaned using the standard RCA cleaning process and used as the back gate of the device. Then, 35-nm Al$_2$O$_3$ and 20-nm ZnO films were deposited successively by atomic layer deposition (ALD) at 250 °C and 200 °C, which served as the blocking layer and CTL of the TFT memory, respectively. It is worth mentioning that the ZnO film has a root–mean–square (RMS) roughness of 0.553 nm. Subsequently, photolithography and wet etching were performed to define the CTL of ZnO. After that, an 8-nm Al$_2$O$_3$ tunneling layer was grown by ALD. The precursors of diethylzinc (DEZ)/H$_2$O and TMA/H$_2$O were used for the growth of ZnO and Al$_2$O$_3$ films, respectively. Thereafter, a 40-nm a-IGZO film was deposited by radio frequency magnetron sputtering as a channel layer at room temperature by using an InGaZnO$_4$ target. The active channel with a width (W)/length (L) of 100/10 μm was then defined by photolithography and diluted HCl etching. Source and drain contacts of Ti/Au (30 nm/70 nm) were formed by e-beam evaporation followed by a lift-off process. Finally, all the fabricated devices were annealed at 250 °C in O$_2$ for 5 min to improve its performance. The electrical characterizations were performed by using a semiconductor parameter analyzer (Agilent B1500A) at room temperature. The threshold voltage (V$_{th}$) is defined as the gate voltage at which the drain current equals to $W/L \times 10^{-9}$ A. The carrier concentration of ZnO films were extracted from Hall effect measurements (Ecopia HMS-3000) at room temperature.

Results and Discussion
Figure 1 shows the schematic diagrams of the fabricated a-IGZO TFT memory device under positive and negative bias programming, respectively. During electrical programming, an electrical pulse is applied on the back gate, and the source and drain electrodes are grounded. Figure 2 shows the programming characteristics of the pristine memory devices under different conditions. For
the pristine memory device, it exhibits an on/off current ratio \( I_{\text{on}}/I_{\text{off}} \) of \( 1.5 \times 10^7 \), field-effect mobility of 7.1 cm\(^2\) V\(^{-1}\) s\(^{-1}\), and a subthreshold swing \( \text{SS} \) of 0.67 V/dec. In terms of 80 ms programming at different positive biases, the \( I_d-V_g \) curve moves gradually in the direction of a positive bias as a function of programming voltage, e.g., the resulting \( V_{\text{th}} \) shift relative to the pristine device (\( \Delta V_{\text{th}} \)) increases from 1.3 to 4.8 V with increasing programming voltage from 8 to 13 V, exhibiting programming saturation at 12 V, as shown in Fig. 2a. Such a significant \( \Delta V_{\text{th}} \) suggests that considerable electrons from the n-type a-IGZO channel are injected into the ZnO CTL. Moreover, when the programming voltage is fixed at 13 V, the \( \Delta V_{\text{th}} \) increases slowly from 2 to 3.1 V with prolonging programming time from 1 \( \mu \)s to 30 ms, as shown in Fig. 2c. Interestingly, when the pristine memory device is programmed at a negative gate bias, the \( V_{\text{th}} \) exhibits a notable shift towards a negative bias, shown in Fig. 2b. For constant programming time of 80 ms, the \( \Delta V_{\text{th}} \) enlarges from -5.2 to -7.4 V with increasing programming bias from -8 to -13 V. Even if the pristine memory device is programmed at -13 V for 1 \( \mu \)s, it can also demonstrate a \( \Delta V_{\text{th}} \) as large as -6.5 V, shown in Fig. 2d. This means that a very large number of electrons are de-trapped from the CTL, hence resulting in remain of plenty of positive charges.

To understand the charge trapping effect of the ZnO layer, a-IGZO TFTs without ZnO CTL are also fabricated as control devices for comparison. Figure 3 shows the
transfer characteristics of the control devices when being programmed under different positive and negative biases, respectively. It is found that the device does not exhibit a discernible $\Delta V_{th}$ regardless of programming voltage polarity and amplitude. This indicates that the aforementioned distinct $\Delta V_{th}$ for the memory devices should be ascribed to the ZnO CTL. On the other hand, it is noted that IGZO is a natural n-type semiconductor, thus electrons in the IGZO channel can be easily injected into the ZnO CTL under a positive gate bias (e.g., +9 V). However, when a negative programming bias is applied to the gate electrode of the device, the a-IGZO channel tends to be depleted, and the hole conduction is hardly achieved [15]. In this case, the device cannot be programmed under negative gate biases, see Fig. 2d. Figure 4 shows the endurance characteristics of the memory as a function of programming/erasing (P/E) cycles. The device exhibits a memory window of 3.7 V for $10^3$ of P/E cycles in the case of 11 V/1 ms programming and −9 V/10 μs erasing. Further, a memory window as large as 7.2 V can be achieved at $10^3$ of P/E cycles with respect to 12 V/1 ms programming and −12 V/10 μs erasing. Table 1 compares the programming and erasing characteristics of various a-IGZO TFT memories [14, 22, 23]. Compared to other devices, our device exhibits a much higher erasing efficiency even under a lower bias (−12 V) and much shorter time (10 μs) in spite of not notable superiority in programming efficiency.

To clarify the origin of electrons de-trapped from the pristine ZnO CTL, various processed ZnO CTLs are compared in the a-IGZO TFT memory devices. Figure 5 shows the programming voltage dependence of $\Delta V_{th}$ for the devices with different ZnO CTLs. It is observed that, for the memory devices with the as-deposited and N$_2$-annealed ZnO CTLs, the resulting $\Delta V_{th}$ exhibits a similar increasing tendency with raising programming voltage despite of voltage polarities. However, for the memory device with the O$_2$-annealed ZnO CTL, the absolute value of $\Delta V_{th}$ shows a significant decrease under the same programming condition, e.g., the absolute value of $\Delta V_{th}$ decreases by 2 and 3 V, respectively, in the case of 13 V/80 ms and −12 V/1 μs programming pulses. Furthermore, saturated programming behaviors are observed for the O$_2$-annealed ZnO CTL in the case of positive and negative gate biases. This should be ascribed to limited traps in the CTL. In a word, the post-annealing in O$_2$ at 250 °C reduces the number of trap centers in the ZnO film, hence leading to a decrease in charge trapping capacity.

To investigate the influence of post-annealing on the properties of the ZnO film, the as-deposited and processed ZnO films are characterized by Hall effect measurements and XPS. As shown in Fig. 6, the ZnO film annealed in N$_2$ at 250 °C shows a carrier concentration of $4.4 \times 10^{19}$ cm$^{-3}$, which is very close to that ($4.5 \times 10^{19}$ cm$^{-3}$) of the as-deposited ZnO film; however, the ZnO film annealed in O$_2$ at 250 °C exhibits a remarkable decrease in carrier concentration, which is equal to $1.8 \times 10^{18}$ cm$^{-3}$. It is reported that the intrinsic donors in n-type ZnO semiconductor films are oxygen vacancies.
Kwon et al. also reported that the O/Zn atomic ratio in the ALD ZnO film decreased gradually from 0.90 to 0.78 with increasing the deposition temperature from 70 to 130 °C [25]. This reveals the existence of oxygen vacancies in ALD ZnO films. Therefore, the O₂-annealing-induced decrease in carrier (electron) concentration should be related to the reduction of oxygen vacancies in the ZnO film. Further, high-resolution O1s XPS spectra of the as-deposited ZnO film and those annealed in N₂ or O₂ are analyzed, as shown in Fig. 7. The deconvoluted three peaks are centered at 530.0, 531.6, and 532.4 eV, corresponding to O²⁻ ions bound with Zn²⁺ (O1), oxygen vacancies (O2), and chemisorbed oxygen element (–OH, etc.) (O3), respectively [26]. Compared with the as-deposited ZnO film, the post-annealing in O₂ generates a decrease of 2.1% in the relative percentage of O₂. Nevertheless, for the ZnO film annealed in N₂, the relative percentage of O₂ is almost unchanged. These results indicate that the O₂ annealing can passivate oxygen vacancies in the ZnO film, but the N₂ annealing cannot do. This further confirms the correlation between oxygen vacancies and carrier concentration.

Based on the aforesaid experimental results, it can be concluded that the programming characteristics of the pristine memory devices are dominated by the concentration of oxygen vacancy-related defects in the ZnO CTL. In other words, oxygen vacancies in the ZnO film primarily serve as trap centers for trapping of positive and negative charges. It

---

**Table 1** Comparisons of the programming and erasing characteristics of various a-IGZO TFT memories with different gate stacks

| Gate stack         | Programming conditions | ΔVₜₚ-P | Erasing conditions | ΔVₜₑ-E | Ref.  |
|--------------------|------------------------|--------|-------------------|--------|-------|
| Al₂O₃/Al₂O₃/Al₂O₃ | Vₔ = 10 V/10 ms         | 2 V    | −10 V/100 s       | −0.2 V | [14]  |
| SiO₂/SmTiOₓ/SiO₂  | Vₔ = 15 V/100 ms       | 2.7 V  | −15 V/100 ms      | −2.2 V | [22]  |
| SiO₂/ErTixOy/SiO₂ | Vₔ = 20V/100 ms        | 3.9 V  | −20 V/100 ms      | −3.9 V | [23]  |
| Al₂O₃/ZnO/Al₂O₃   | Vₔ = 13 V/1 μs         | 2 V    | −12 V/10 μs       | −7.2 V | This work |

ΔVₜₚ-P means the threshold voltage shift relative to the pristine device after programming and ΔVₜₑ-E implies the threshold voltage shift relative to the programmed device after erasing.
is reported that oxygen vacancy-related defects in ZnO include neutral oxygen vacancy (V\(_O^0\)), singly ionized oxygen vacancy (V\(_O^+\)), and doubly ionized oxygen vacancy (V\(_O^{2+}\)), whose energy levels are located at 0.02–0.04, 0.3–0.45, and 0.61 eV, respectively, below the conduction band minimum of ZnO [27, 28]. Since the as-deposited ZnO film shows a high electron concentration in our case, the concentration of neutral oxygen vacancies serving as shallow donors should be much higher than that of ionized oxygen vacancies (V\(_O^+\) and V\(_O^{2+}\)). In terms of programming at a positive gate bias, electrons in the accumulation layer of the a-IGZO channel are injected into the ZnO layer by the Fowler–Nordheim (F-N) tunneling mechanism, which is demonstrated by an incremental \(\Delta V_{th}\) with enhancing programming voltage in Fig. 2a. Meanwhile, it is expected that these electrons are trapped preferentially at deep levels of V\(_O^+\) and V\(_O^{2+}\), as depicted in Fig. 8a. This causes a shift of \(V_{th}\) towards a positive bias. Of course, in addition to oxygen vacancies that trap electrons, other defects also could capture electrons. However, our experimental data indicate that oxygen vacancies play a crucial role in electron trapping as well as positive charge capture, as revealed in Fig. 5. Under negative programming voltage, the neutral oxygen vacancies in the pristine ZnO CTL dominantly donate electrons because of the shallowest energy level [27, 28], and the released electrons tunnel from the ZnO CTL into the channel, hence leading to the formation of positively charged oxygen vacancies (e.g., V\(_O^{+}\)), as shown in Fig. 8b. This causes a shift of \(V_{th}\) in the direction of negative bias, as indicated in Fig. 2b. Further, owing to a higher concentration of neutral oxygen vacancies (V\(_O^0\)) in the as-deposited CTL of ZnO, the pristine memory device exhibits a much higher programming efficiency under the negative gate bias than under the positive gate bias. For example, the absolute value of \(\Delta V_{th}\) is as large as 6.5 V after programming at –13
V for 1 μs (see Fig. 2d); however, the ΔV_th is equal to 2 V after programming at 13 V for 1 μs (Fig. 2c). This is because the former is determined mainly by the concentration of VO, and the latter is dominated by the concentrations of VO+ and VO2+.

**Conclusions**

In summary, we fabricated a bipolar programmable a-IGZO TFT memory with an atomic-layer-deposited ZnO CTL. Compared with the programming under a positive gate bias, the programming under a negative gate bias can generate a much higher efficiency. This is because different oxygen vacancy defects take effect during voltage-polarity-dependent programming. That is, deep defects of VO+ and VO2+ play a key role for electrons trapping during positive bias programming, and shallow defects of VO mainly donate electrons during negative bias programming, resulting in the generation of positively charged oxygen vacancies.
Abbreviations
a-IGZO: Amorphous indium-gallium-zinc-oxide; ALD: Atomic layer deposition; CTL: Charge trapping layer; TFT: Thin-film transistor; XPS: X-ray photoelectron spectroscopy

Acknowledgements
The authors would like to acknowledge the financial support by the National Natural Science Foundation of China (grant nos. 61874029, 61774041) and National Key Technologies Research and Development Program of China (grant no. 2015ZX02102-003).

Authors’ Contributions
DDL performed the experiment, data processing, and manuscript drafting. SJD and WJL modified the manuscript. Other authors help review and discuss the manuscript. All authors read and approved the final manuscript.

Availability of Data and Materials
The datasets supporting the conclusions of this manuscript are included within the manuscript.

Competing Interests
The authors declare that they have no competing interests.

Received: 21 June 2019 Accepted: 6 November 2019
Published online: 02 December 2019

References
1. Iwasaki T, Itagaki N, Den T, Kumomi H, Nomura K, Kamiya T, Hosono H (2007) Combinatorial approach to thin-film transistors using multicomponent semiconductor channels: an application to amorphous oxide semiconductors in In-Ga-Zn-O system. Appl Phys Lett 90:488
2. Jang J, Park JC, Kong D, Kim DM, Lee JS, Sohn BH, Cho IH, Kim DH (2011) Endurance characteristics of amorphous-InGaZnO transparent flash memory with gold nanocrystal storage layer. IEEE Electron Device Letters 32:3940–3947
3. Suresh A, Novak S, Welleuens P, Misra V, Muth JF (2009) Transparent indium gallium zinc oxide transistor based floating gate memory with platinum nanoparticles in the gate dielectric. Appl Phys Lett 94:378
4. Chen WT, Zan HW (2012) High-performance light-erasable memory and real-time ultraviolet detector based on unannealed indium-gallium-zinc-oxide thin-film transistor. IEEE Electron Device Letter 33:77–79
5. Zielewski SJ, Zhou Z, Li FZ, Kang XL, Meng Y, Ho J, Kudrawiec R (2019) Optical properties of InGa2O3 nanowires revealed by photoacoustic spectroscopy. ACS Appl Mater Interfaces 11:19260–19266
6. Li FZ, Yip SP, Dong R, Zhou ZY, Lan CY, Liang XG, Li DP, Meng Y, Kang XL, Ho JC (2019) Crystalline InGaZnO quaternary nanowires with superlattice structure for high-performance thin-film transistors. Nano Res 12:1796–1803
7. Song LF, Luo LQ, Li X, Liu D, Han N, Liu L, Qiu YB, Ho JC, Wang FY (2019) Modulating electrical performances of InOx nanofiber channel thin film transistors via Sr doping. Adv Electron Mater 5:1800707
8. Fang FY, Song LF, Zhang HC, Meng Y, Luo LQ, Xi Y, Liu L, Han N, Yang ZX, Tang J, Shan FK, Ho JC (2018) ZnO nanofiber thin-film transistors with low-operating voltages. Adv Electron Mater 4:1700336
9. Zhang JC, Meng Y, Song LF, Luo LQ, Qin YB, Han N, Yang ZX, Liu L, Ho JC, Wang YF (2018) High-performance enhancement-mode thin-film transistors based on Mg-doped InOx nanofiber networks. Nano Res 11:1227–1237
10. Zheng L, Cheng XH, Cao D, Wang G, Wang ZJ, Xu DW, Xie C, Shen LY, Yu YH, Shen DS (2014) Improvement of ALOx films on graphene grown by atomic layer deposition with Pre-H2O treatment. ACS Appl Mater Interfaces 6:7014–2019
11. Zheng L, Zhou WJ, Wang G, Cheng XH, Hu WD, Zhou W, Liu ZD, Yang SW, Xu KM, Luo M, Yu YH (2018) Ambipolar graphene-quantum dot phototransistors with CMOS compatibility. ACS Appl Mater Interfaces 9:1800985
12. Zheng L, Cheng XH, Yu YH, Xie YH, Li XH, Wang ZJ (2015) Controlled direct growth of Al2O3-doped HFO2 films on graphene by H2O-based atomic layer deposition. Phys Chem Chem Phys 17:3179–3185
13. Nomura K, Kamiya T, Yanagi H, Ikemaga E, Yang K, Kobayashi K, Hirano M, Hosono H (2008) Subgap states in transparent amorphous oxide semiconductor, In-Ga-Zn-O, observed by bulk sensitive x-ray photoelectron spectroscopy. Appl Phys Lett 92:334
14. Li Y, Pei YL, Hu RQ, Chen ZM, Ni YQ, Lin JY, Chen YT, Zhang XK, Shen Z, Liang J, Wang G, Duan H (2015) Charge trapping memory characteristics of amorphous-indium-gallium-zinc oxide thin-film transistors with defect-engineered alumina dielectric. IEEE Trans Electron Devices 62:1184–1188
15. Chen S, Zhang WP, Cui XM, Ding SH, Sun QQ, Zhang DW (2014) Monochromatic light-assisted erasing effects of In-Ga-Zn-O thin film transistor memory with Al2O3/Zn-doped Al2O3/O2 stacks. Appl Phys Lett 104:103504
16. Cui XM, Chen S, Ding SJ, Sun QQ (2017) Unique UV-erasable In-Ga-Zn-O TFT memory with self-assembled Pt nanocrystals. IEEE Electron Device Lett 38:1011–1013
17. Qian SB, Shao Y, Liu WJ, Zhang DW, Ding SJ (2017) Erasing-modes dependent performance of a-KZTO TFT memory with atomic-layer-deposited Ni nanocrystal charge storage layer. IEEE Trans Electron Devices 64:3023–3027
18. Yun DJ, Kang HB, Yoon SM (2016) Process optimization and device characterization of nonvolatile charge trap memory transistors using In-Ga-Zn-O thin films as both charge trap and active channel layers. IEEE Trans Electron Devices 63:3128–3134
19. Kim SJ, Park MJ, Yun DJ, Lee WH, Kim GH, Yoon SM (2016) High performance and stable flexible memory thin-film transistors using In-Ga-Zn-O channel and ZnO charge-trap layers on poly(ethylene naphthalate) substrate. IEEE Trans Electron Devices 63:1557–1564
20. Bak JY, Ryu MK, Park SHK, Hwang CS, Yoon SJ (2014) Impact of charge-trap layer conductivity control on device performances of top-gate memory thin-film transistors using IZO channel and ZnO charge-trap layer. IEEE Trans Electron Devices 61:2404–2411
21. Zhang WP, Qian SB, Liu WJ, Ding SJ, Zhang DW (2015) Novel multi-level cell TFT memory with an In-Ga-Zn-O charge storage layer and channel. IEEE Trans Electron Devices 62:1021–1023
22. Her JL, Chen FH, Chen CH, Pan TM (2015) Electrical characteristics of gallium-indium-zinc oxide thin-film transistor nonvolatile memory with Sm2O3 and SmTiO3 charge trapping layers. RSC Advances 5:8566–8570
23. Pan TM, Chen CH, Hu YH, Wang HC, Her KJ (2016) Comparison of structural and electrical properties of Er2O3 and ErTi2O3 charge-trapping layers for InGaZnO thin-film transistor nonvolatile memory devices. IEEE Electron Device Lett 37:179–181
24. Mahan GD (1983) Intrinsics defects in ZnO varistors. J Appl Phys 54:3825–3832
25. Kwon S, Bang S, Lee S, Jeon S, Jeong W, Kim H, Gong SC, Chang HJ, Park H, Jeon H (2009) Characteristics of the ZnO thin film transistor by atomic layer deposition at various temperatures. Semicond Sci Technol 24:035015
26. Wang YH, Ma Q, Zheng LL, Liu WJ, Ding SJ, Lu HL, Zhang DW (2016) Performance improvement of atomic layer-deposited ZnO/Al2O3 thin-film transistors by low-temperature annealing in air. IEEE Trans Electron Devices 63:1893–1898
27. Pöppel A, Völkel G (1991) ESR and Photo-ESR investigations of zinc vacancies and interstitial oxygen ions in undoped ZnO ceramics. Phys Status Solidi A 118:217–233
28. Erhart P, Albe K, Klein A (2006) First-principles study of intrinsic point defects in ZnO: role of band structure, volume relaxation, and finite-size effects. Phys Rev B 73:205203

Publisher’s Note
Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.