Monte Carlo Simulations of Off-Lattice Polymers

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Abstract

We point out that a newly introduced recursive algorithm for lattice polymers has a much wider range of applicability. In particular, we apply it to the simulation of off-lattice polymers with Lennard-Jones potentials between non-bonded monomers and either delta or harmonic potentials between bonded monomers. Our algorithm allows particularly easy calculations of the free energy, and seems in general more efficient than other existing algorithms.
1 Introduction

In a number of recent papers, new Monte Carlo schemes for simulating off-lattice polymers have been proposed [1]–[8]. A particular aspect there was the calculation of chemical potentials. This is not easy in most schemes, which has led to some controversy [9, 10]. In this note we want to point out that a recently introduced recursive implementation of the enrichment method for lattice polymers [11] can be easily adapted to this problem. It is both easy to implement and efficient, and the computation of the chemical potential is straightforward. Indeed, for polymers without long-range monomer-monomer interactions and without interactions with any solvent our method seems to be faster than all methods mentioned above. It is not efficient for systems at extremely low energies (when Boltzmann factors due to attractive potentials between individual monomers become > 10), and for systems with long-range forces [12, 13].

Though we shall apply our algorithm only to polymer systems, we should point that it is much more general. It can be applied to any equilibrium system which can be broken up into discrete units, labeled by an index $i = 1, \ldots, N$, and whose internal energy can be written as

$$U = \sum_{i=1}^{N} U_i,$$

(1)

where $U_i$ depends only on units with label $i' \leq i$. In the case of a polymer with potential $u_{ij}$ between non-bonded monomers and potential $v_i$ between monomers $i$ and $i - 1$, we choose of course $U_i = v_i + \sum_{j<i} u_{ij}$ for $i > 1$, while $U_1 = v_1$ is evaluated with $x_0 = 0$. Thus the start of polymer chain is anchored to $x = 0$.

The algorithm basically tries to build the system by assembling it unit by unit. To obtain the correct Boltzmann weights, the entire assembled configuration has to be discarded occasionally (with probabilities depending on these weights) if the following units do not fit. A typical example is a self-avoiding walk on a lattice, where the configuration has to be discarded if the next step would lead to a self intersection. In order to compensate for this “attrition”, we use an “enrichment” method, the basic idea of which was already introduced more than 30 years ago [14]. Instead of trying just one continuation from a partially assembled system, the intermediate sample is “enriched” by replicas which serve as starting points for independent continuations. Thus from each partly assembled system more than one continuation is attempted. These attempts are made irrespective of whether one of them is successful or not, which leads immediately to an unbiased Gibbs ensemble and which distinguished the method from most other proposals to overcome attrition.

But in contrast to earlier implementations of the enrichment idea which were of “breadth first” type [15, 16, 17, 18], our implementation is “depth first” [19]. This implies completely different data structures. In particular, it means that a partly assembled system can reside in the fast memory of the computer even in very large simulations. This avoids the very time consuming data transfer needed in breadth-first implementations, unless the systems are very small. As discussed in [17], simulating a system with $N$ units one has to simulate $>> N$ ensemble members
simultaneously in a breadth first algorithm. This requires a storage space > \( O(N^2) \), which limits severely the system sizes. But even if the entire ensemble fits into memory, a breadth first algorithm is slower by a factor \( O(N) \) since adding a new unit takes a time \( O(N) \) \(^{[17]} \) (there is a finite probability that a new replica has to be created) while it needs only a time \( O(1) \) for a depth first algorithm. Also, the simplest and most intuitive implementation of a depth first algorithm is via recursive function calls. In this case the compiler makes all the book-keeping which is fast but quite non-trivial in this approach. Thus our method avoids all problems which have made the enrichment method unpopular in the past.

The only disadvantage of a depth first implementation is that we have to guess the attrition in advance. In a breadth-first approach, level \( i \) is treated only after all previous levels have been finished, and thus the attrition on the previous levels is known. If we know the amount of attrition sufficiently well from other sources, we do not have any problem in a depth first approach either. Otherwise, the best strategy is to start with small systems and a conservative estimate of the attrition, and to increase the system size in separate runs as the attrition gets better and better known \(^{[20]} \).

\section{Algorithm}

Our aim is to compute the partition function

\[ Z_N = \int d\mathbf{x}_1 \ldots d\mathbf{x}_N e^{-\beta U(\mathbf{x}_1 \ldots \mathbf{x}_N)}, \quad \beta = \left( k_B T \right)^{-1}. \]  

(2)

In addition, we consider also “partial partition functions” which describe the last \( N - i \) units in the static background field created by the first \( i + 1 \) units,

\[ Z_{N-i|i}(\mathbf{x}_1 \ldots \mathbf{x}_i) = \int d\mathbf{x}_{i+1} \ldots d\mathbf{x}_N e^{-\beta \sum_{j=i+1}^{N} U_j(\mathbf{x}_1 \ldots \mathbf{x}_j)}. \]

(3)

They can be written recursively,

\[ Z_{N-i+1|i-1}(\mathbf{x}_1, \ldots, \mathbf{x}_{i-1}) = \int d\mathbf{x}_i \, e^{-\beta U_i(\mathbf{x}_1 \ldots \mathbf{x}_i)} \, Z_{N-i|i}(\mathbf{x}_1, \ldots, \mathbf{x}_i) \]

(4)

with \( Z_{0|N}(\mathbf{x}_1, \ldots, \mathbf{x}_N) \equiv 1 \) and \( Z_{N|0} \equiv Z_N \). The basic strategy will be to compute Monte Carlo estimates for the partial partition functions using this recursion relation (this will be done implicitly, and the explicit code needed to do it will be very compact). The total partition function is generated by “assembling” the last units first (which means just summing over suitable statistical samples), and working one’s way back. The task is completed when finally the sample points for the first unit are summed over. With this strategy in mind we will in the following concentrate on one typical step in the recursion relation where \( Z_{N-i|i}(\mathbf{x}_1 \ldots \mathbf{x}_i) \) is assumed to be known, and \( Z_{N-i+1|i-1}(\mathbf{x}_1 \ldots \mathbf{x}_{i-1}) \) is to be computed.

We assume that the potential \( U_i \) can be split into two parts,

\[ U_i(\mathbf{x}_1 \ldots \mathbf{x}_i) = U_i^{(0)}(\mathbf{x}_1 \ldots \mathbf{x}_i) + \Delta U_i(\mathbf{x}_1 \ldots \mathbf{x}_i), \]

(5)
with the following properties:

(i) The partial partition functions $Z_{N-i|i}$ associated with $U^{(0)}$ can be computed for each background configuration $(x_1 \ldots x_i)$ either analytically or by some other method which is independent of the present Monte Carlo method;

(ii) A fast (pseudo-)random number generator exists which produces points $\xi_k$ distributed with the density

$$
\rho_i^{(0)}(\xi|x_1 \ldots x_{i-1}) = \frac{Z_{N-i|i}^{(0)}(x_1 \ldots x_{i-1}, \xi)}{Z_{N-i+1|i-1}^{(0)}(x_1 \ldots x_{i-1})} e^{-\beta U_i^{(0)}(x_1 \ldots x_{i-1}, \xi)}
$$

(notice that this is normalized due to eq. (11));

(iii) $\Delta U_i$ is “kind” in the sense that the integral $\int d\xi \rho_i^{(0)} e^{-\beta \Delta U_i}$ converges for all background configurations and is not too big. We should stress that the last condition affects only the efficiency of the method, but is not related to any bias. In particular, we make no series expansion or truncation in higher powers of $\Delta U_i$, or anything of that sort. In general, it will be sufficient if $U_i^{(0)}$ has correct asymptotic behavior for the integral to converge, and has roughly the same shape as $U_i$.

Our $U^{(0)}$ is similar to the “guiding field” in [16, 17].

Using this decomposition of $U_i$ one shows easily that

$$
Z_{N-i+1|i-1}(x_1 \ldots x_{i-1}) = Z_{N-i+1|i-1}^{(0)}(x_1 \ldots x_{i-1}) \int d\xi \rho_i^{(0)}(\xi|x_1 \ldots x_{i-1})
$$

$$
e^{-\beta \Delta U_i(x_1 \ldots x_{i-1}, \xi)} \frac{Z_{N-i|i}^{(0)}(x_1 \ldots x_{i-1}, \xi)}{Z_{N-i+1|i-1}^{(0)}(x_1 \ldots x_{i-1})}.
$$

The integral over $\xi$ can now be approximated by a sum over random points $\xi_k$ obtained by means of the above random number generator, and we obtain

$$
Z_{N-i+1|i-1}(x_1 \ldots x_{i-1}) = Z_{N-i+1|i-1}^{(0)} \lim_{M \to \infty} \frac{1}{M} \sum_{k=1}^{M}
$$

$$
e^{-\beta \Delta U_i(x_1 \ldots x_{i-1}, \xi_k)} \frac{Z_{N-i|i}^{(0)}(x_1 \ldots x_{i-1}, \xi_k)}{Z_{N-i+1|i-1}^{(0)}(x_1 \ldots x_{i-1})}.
$$

Assume that we have already an estimator for $Z_{N-i|i}$. Then an estimator for $Z_{N-i+1|i-1}$ is obtained by either associating a weight

$$
w_i(\xi_k; x_1 \ldots x_{i-1}) = \frac{1}{M} \frac{Z_{N-i+1|i-1}^{(0)}(x_1 \ldots x_{i-1})}{Z_{N-i|i}^{(0)}(x_1 \ldots x_{i-1}, \xi_k)} e^{-\beta \Delta U_i(x_1 \ldots x_{i-1}, \xi_k)}
$$

with each $\xi_k$, or — and this is the method used in our approach — by replacing each $\xi_k$ in the average by $p_i w_i$ replicas of itself, each counted with weight 1 and labeled by an index $\alpha$. Here $p_i$ is a parameter (independent of $x_i$ and $\xi_k$) which is in principle arbitrary (more about it will be said below) and which controls the size of
the sample by counterbalancing the attrition during the step $i \to i - 1$. This gives then our MC estimate

$$Z_{N-i+1|i-1}(x_1 \ldots x_{i-1}) \approx \frac{1}{p_i} \sum_{k=1}^{M} \sum_{\text{replicas } \alpha} Z_{N-i+1|i-1}^{[\alpha]}(x_1 \ldots x_{i-1}, \xi_k). \quad (12)$$

The superscript $\alpha$ on the partial partition function on the rhs. is to indicate that we use of course different sample points ($x_{i+1} \ldots x_N$) for each replica, even though the backgrounds are the same.

The parameter $p_i$ has to be chosen carefully: large $p_i$ will lead to samples whose sizes increase quickly with $i$, leading to excessive CPU times for large $N$, while small $p_i$ lead to too small samples for large $i$. Optimally, $p_i$ should be chosen such that the sample size is roughly independent of $i$. Notice that this means in particular that $M$ will not be large. Indeed, most numerical results quoted below are obtained with $M = 1$. Large statistics is not obtained by trying many different continuations of each partially built chain, but by making many independent runs. It is only for extremely low temperatures (not studied here) that $M \gg 1$ should be advantageous since it allows a more uniform covering of configuration space.

We should point out that the above two possibilities (of using $w_i$ either as a weight or as a multiplicity of replicas) are indeed just two special cases within a much wider range of possible choices. They are all distinguished by a different balance between equidistribution of the statistical sample and equal weights put on all sample point. It might well be that for different purposes different variants are optimal, but we shall in the following discuss only the choice of uniform weights, corresponding to perfect importance sampling. It has the advantage that all thermal averages are just normal averages without any additional weight factors except for the weights $p_i$. In particular, the incremental chemical potential is simply

$$\mu_i = -k_B T \log \frac{Z_i}{Z_{i-1}} \approx -k_B T \log \frac{m_i}{m_{i-1} p_i} \quad (13)$$

where $m_i$ is the total number of sample point replicas at level $i$ (i.e., the total number of subroutine calls at depth $i$ in the algorithm described below).

Technically, our method is implemented by means of a recursively called subroutine which has the level $i$ as argument. Basically, it just chooses a random point $\xi$, inserts a monomer at this point and computes its weight $w_i(\xi)$, and makes in the average $p_i w_i(\xi)$ calls to itself if $i < N$, with new argument $i + 1$. After returning from all these subroutine calls, the monomer at $\xi$ is removed and the subroutine is left. This is of course complemented by updating the statistics for whatever observable is to be measured.

Finally, we have to specify what we mean by “make ... calls in average”. In principle, we can choose any distribution for the number of calls, provided it gives the right average. But efficiency will depend on this choice. One possibility would

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1We use the word “attrition” for conformity with the use in the literature on self-avoiding walks. It should be noted that in our case it does not necessarily imply a depletion of chains, but can also imply the opposite, depending on the sign of the potential.
be e.g. a Poissonian (this would be in spirit with the first application of this method in a lattice model \[20\]), but in general this is not the best choice. As in lattice models, it seems that the optimal choice depends on the specific situation, and in extreme cases (steep potentials and low temperatures) some experimentation will be needed. As a rule of thumb we propose to choose the distribution such that it has the smallest possible variance \[11\]. Thus, if \( p_i \) \( w_i(\xi) = k + \eta \) with \( \eta \in [0, 1) \) and \( k \) integer, we first make \( k \) calls and then add one more call with probability \( \eta \).

### 3 Applications

We applied this to two versions of a model where non-bonded monomers interact by Lennard-Jones potentials in 3-d space,

\[
 u_{ij} = 4[r_{ij}^{-12} - r_{ij}^{-6}], \quad r_{ij} = |\mathbf{r}_{ij}| = |\mathbf{x}_i - \mathbf{x}_j|. \tag{14}
\]

In the first version \[3\], the force between bonded monomers is harmonic,

\[
 v_i = \frac{\kappa}{2} (r_{i,i-1} - 1)^2, \quad \kappa = 400 \tag{15}
\]

for \( r_{i,i-1} > 0.5 \). For \( r_{i,i-1} < 0.5 \), the potential is infinite. In the second version \[4\], it is simply provided by hard rods which keep a fixed distance \( r_{i,i-1} = 1 \). Notice that we did not truncate the potential at large distances, in contrast to previous studies \[1, 7\]. As far as we can see, the approximations involved in correcting for this truncation should be the only possible source for eventual disagreements with these works.

In the second version, a natural choice of \( U^{(0)}_i \) is given by an isotropic delta potential\[4\] fixing \( r_{i,i-1} \) at 1. Thus the vectors \( \xi \) were chosen randomly on the surface of a sphere centered at \( x_{i-1} \). The corresponding partial partition functions are \( Z^{(0)}_{n|i} = Z^{(0)}_{n} = (4\pi)^n \).

In the first version the choice is less obvious since it is not so easy to produce random points according to \( e^{-\beta v_i} \) with \( v_i \) given by eq.(15). We thus took

\[
 U^{(0)}_i = v_i(r_{i,i-1}) + k_B T \ln(r_{i,i-1}^2) \tag{16}
\]

For this choice the radial distribution function is a Gaussian centered at \( r = 1 \), \( \rho^{(0)}(r_{i,i-1}) \propto r_{i,i-1}^{-2} e^{-\beta v_i} \), and \( Z^{(0)}_n = [(2\pi)^3/(\kappa\beta)]^{n/2} \) for \( \kappa\beta >> 1 \)[5].

Fig.[2] shows our data for the excess chemical potential \( \beta \mu^{ex} \) for the hard rod and the harmonic potential, respectively. The value \( \beta = 1/1.2 \) was chosen to compare our data to those of \[7\]. From fig.5 of that paper we see that \( \beta \mu^{ex} = -10.4 \pm 0.4 \) for a hard rod chain with \( N = 30 \) monomers. This is much smaller (in absolute value) than our estimate \( \beta \mu^{ex} = -15.35 \pm 0.05 \). Part of this discrepancy can be explained

\[\text{2By this we mean of course not strictly a delta function but a potential which is enough singular to give a delta function for the equilibrium density, } e^{-\beta U^{(0)}_i} = \delta(r_{i,i-1} - 1).\]

\[\text{3If this condition is not fulfilled we get an additional term proportional to an error function due to the centering of } r_{i,i-1} \text{ around } 1.\]
Figure 1: Results for the excess chemical potential at $T = 1/\beta = 1.2$. The upper curve is related to the hard rod potential, while the lower one is the result for the harmonic potential of eq.(15). Statistical errors are smaller than the thickness of the line.

by the fact that the Lennard-Jones potential was truncated in [7] at $r = r_c \equiv 2.5$ and its contribution from $r > r_c$ was estimated analytically. This was not done in our simulations, where all potentials were taken into account exactly. But by performing the same truncation as in [7] without correcting for it at all, we estimated that even this is much too small ($\approx 9 - 10\%$) to explain the discrepancy. The simulations of [1] have too large statistical errors for a similarly detailed comparison.

Averages of the squared end–to–end distance for longer chains are shown in fig. 2. They clearly indicate that $\beta = 1/1.2$ is far below the $\Theta$–temperature in agreement with the fact that $\mu^{\text{ex}}$ is negative and decreasing with $N$. In fig. 3 we show our data for $\beta \mu^{\text{inc}} = \beta(\mu^{\text{ex}}_N - \mu^{\text{ex}}_{N-1})$, which represents the free energy needed to add one more monomer to the chain. At the $\Theta$–point we expect $\mu^{\text{inc}}$ to be independent on $N$. The chains are still too short to pin down $\beta_\Theta$ exactly, but the plots unambiguously show that $\beta_\Theta$ is much smaller then 0.36, the value given in [2]. Together with the data from fig. 2 we would say that $0.2 \leq \beta_\Theta \leq 0.23$. Both, the data of fig. 3 and that of fig. 2 were produced using the hard rod potential for bonded monomers and the full LJ potential for nonbonded monomers, i.e. without performing any cutoff at large $r_{ij}$. Each data set is based on at least $10^6$ ‘tours’. We call a tour the set of all (correlated) chains produced by the same initial subroutine call from the main
routine.

Figure 2: Simulation results for the average squared end–to–end distance for $\beta = 0.175, \ldots, 0.375$ with $\Delta \beta = 0.025$ and additionally for $\beta = 0.4, 0.6, 1/1.2$. The data clearly show that the chains are already collapsed at $\beta \geq 0.25$.

4 Generalizations and Outlook

In the above applications, we did not truncate the potential at large $r$. Thus, inserting a new monomer takes a time $O(N)$. For larger $N$ this is no longer tolerable. If the potential is truncated in such a case, one should also use efficient data structures for searching relevant neighbors [21]. With this we can achieve $O(1)$ behavior.

Our algorithm as described above can become inefficient for two main reasons, but in both cases this can be cured by minor modifications: first of all, if the temperature is very low, the Boltzmann factors $e^{-\beta \Delta U_i}$ for single monomers can become very large. The efficiency of the method results from the fact that large Boltzmann factors for the entire chain are split into smaller factors for individual monomers. If the latter factors themselves become large and are spread over a large interval, then most random positions will either be immediately discarded since they have very small $w_i$, or they will have very large $w_i$ and lead thus to very many replicas and correspondingly to huge fluctuations. A case where this made our method quite unsuccessful is the random heteropolymer model of [22].
Figure 3: Simulation results for $\beta \mu_{\text{inc}}$ for $\beta = 0.175, \ldots, 0.375$ with $\Delta \beta = 0.025$. One can see that $\mu_{\text{inc}}$ is independent of $N$ for $\beta \approx 0.2$ to 0.225. So we can expect the $\Theta$–point somewhere between these two values.

The simplest way out of this dilemma consists in choosing $M >> 1$. In this way we will have even more points which are essentially useless since they have very small $w_i$. But the regions in state space with high probability will be sampled more evenly, and this should be more important. Another popular remedy consists in making the bulk of the simulations at higher temperatures, and quenching down to the desired temperatures in regular time intervals [18].

The other case where the above algorithm gives poor results is provided by systems in which originally favored configurations lead finally to dead ends, while originally unfavored configurations become more important as the growth of the system continues. Two specific examples are dense polymer systems in 2-d [6] and polymers with long range repulsive forces [12, 13].

Assume we want to grow a long self avoiding walk in a finite 2-d region such that it fills a large fraction of the area. As we place the first monomers, all configurations are equally likely. But those which leave large closed voids all are dead ends since the walk later on cannot penetrate into the voids. Obviously it would be much better in this case to bias the walks against large voids from the very beginning.

The situation is similar for a polymer with repulsive Coulomb potentials. There, unless the force is very strong, the effect of the repulsion is not too strong for end monomers, and hence new monomers are added without a strong radial bias. But the stretching force on a monomer deep inside the chain is much bigger (since all forces essentially add up), and the configuration is much more stretched inside the chain. Thus, when an existing chain is to be prolonged, most of the existing configurations
have to be discarded, in order to be replaced by stretched configurations which at first (when they are assembled) are very improbable.

As we said, a way out of this problem is to use biased walks. This means that the number of replicas is not strictly proportional to \( w \), but is larger in those regions which we suspect to become more important later. Of course it means also that we have to replace eq. (12) by a weighted sum. It is e.g. known that the Rosenbluth trick [23] leads to a bias towards more compact SAW’s. We found indeed that our method with Rosenbluth weights instead of uniform weights was more efficient in giving SAW’s which fill a square with periodic boundary conditions, but it leads to much larger statistical fluctuations at low density. We should point out that this possibility of biasing is independent of the choice of \( U^{(0)} \), something which seems to have been missed in [16, 17, 18].

Finally, we tried our method also for non-polymeric systems. For instance, we simulated the 2-d Ising model with spins numbered in the same way as they would be e.g. stored in a FORTRAN array. Though the method worked decently, it could not compete either with cluster flip methods (due to their much more efficient moves) nor with conventional Monte Carlo schemes which can be made very efficient by vectorization and multi-spin coding. We should mention that the possible application spin models, and to the Ising model in particular, was also pointed out in [24] in the context of the breadth-first approach.

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