Water mediated dielectric polarizability and electron charge transport properties of high resistance natural fibers

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Recent studies showed that silk and human hair fibers develop thermoelectric properties at optimal water, temperature and light conditions. The nature of charge carriers and the role of water in mediating charge conduction in these fibers is an unexplored issue. By studying four different classes of natural fibers, viz., silk cocoon, human hair, jute and corn silk, we uncover their common electrical transport properties and its dependence on water concentration and temperature. All these fibers uniformly exhibit nonlinear, hysteretic current - voltage characteristics, which scale with water concentration. The optimal electrical conductivity shows thermally activated hopping transport mechanism. Scanning tunneling microscope (STM) and dielectric measurements of silk cocoon fibers showed the electronic density of states and dielectric properties of the hydrated medium enhances with water concentration. Electron paramagnetic resonance (EPR) study reveals that the charge carriers in these membranes are electronic in nature. Our results are explained through the mechanism of hopping of a Polaron, which is an electron surrounded by positive charge fluctuations created by water molecules. The mechanism unravels the peculiar role water plays in mediating electrical activity in these membranes and also opens the possibility for exploring such charge transport mechanism in other biological membranes.

Recent studies show that silk and human hair fibers have novel thermoelectric properties1,2. Understanding the electrical properties of hydrated biological membranes is important not only for applications but also for its fundamental relevance in investigating the behavior of electrical circuitry in the natural living world. In biological membranes charge carriers can be either electrons or protons3 and charge conduction occurs in an aqueous medium. One may ask if water is necessary for charge transport in such systems? It is well known that water is a vital ingredient in sustaining biological and chemical functions in plant and animal tissues4–6. Water is important to maintain protein functionality7, regulating their shape8 and mobility9,10. Water also affects protonic charge transport across proteins. Studies suggest that the transport of protons across biological macromolecules4,7,11 like proteins occur via the formation of an intermediate ionic complex of water molecules (e.g. (H2O)n+ species)11–21.

The charges are considered to hop along proton wires created on the hydrated molecular backbone of biological macromolecules. While the role of water, in mediating proton charge transport has been widely investigated, its role in promoting electron charge transport in such systems has not been well explored.

The recent experiments on the most abundant natural fibers from insect, animal and plant kingdom22–26 viz., silk cocoon (Bombyx mori, Antheraea mylitta), human hair, jute and corn-silk (obtained from the corn cob of the maize plant) show that at optimum moisture, temperature and light conditions, these fibers exhibit thermoelectric, photo-electric and magnetic properties1,2,27,28. However, the nature of charge carriers and the effect of water (if any) on the charge transport properties of these fibers, have not been explored. To address the above questions,

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we study the electrical properties of four natural fibers, viz., silk cocoon, human hair, jute and corn-silk. These fibers represent four distinct class of chemical families of fibers\textsuperscript{22–26}, viz., silk cocoon is a beta-sheet fibroin protein forming a composite structure by adhering with a glue like protein viz., sericin\textsuperscript{22,23}, human hair represents alpha-helical protein\textsuperscript{24}, jute represents a pure carbohydra and protein hybrid\textsuperscript{25} (see Methods and also the section: Image of samples with electrical contacts, in Supplementary Information). In our measurements, the fibers are maintained in their native state without denaturing them in any way. All membranes exhibit low electrical conductivity, which improves with increasing water content. The membranes have a nonlinear, hysteretic current (I) - voltage (V) characteristics, where the hysteresis scales with the water content. Temperature dependent electrical conductivity reveals a thermally activated charge transport mechanism in these natural membranes. Our scanning tunneling microscope (STM) measurements show strong water induced enhancement in the electronic density of states and the development of a local internal electric field mechanism in these natural membranes. Our scanning tunneling microscope (STM) measurements show strong electronic nature of the charge carriers in these membranes and also the presence of water weakens the coupling of the charges with their complex chemical environment. We model the charge transport in these hydrated dielectric membranes occurs via the model of hopping of a Polaron complex.

**Results**

**Bulk electrical characterization of the biological membranes using transport measurement.** Here we study the bulk electrical conducting properties of natural membranes made from a collection of *Bombyx mori* silk cocoon (SCM\textsubscript{BMW}) fibers (obtained from the silk cocoon). The electrical properties of these fibers are measured by making traditional electrical contacts directly on the membrane (see Methods). The fibers are maintained in their native state and all the measurements are performed under controlled hydration levels (d). The dry silk cocoon membrane is an insulating material with a resistance \( R > 50 \text{ M}\Omega \). Figure 1a shows that when the water content in the membrane is above a critical value \( d_{c} \sim 0.1 \text{ mg mm}^{-3} \), the resistance \( R \) appears to be almost uniform. Note that below \( d_{c} \) the membrane resistance diverges rapidly (Fig. 1a).

The inset of Fig. 1a shows that the water content \( (d) \) of the membrane depletes linearly with time \( (t) \). The value of \( d \) at \( t = 0 \), corresponds to the situation when the membrane is saturated with water. For a time interval of 100 minutes from the start of the experiment when the membrane is dipped in water and lifted, the wet silk cocoon membrane exhibits a linear \( d(t) \) relationship. However, at small \( d \) values the \( d(t) \) asymptotically approaches zero with a nonlinear tail like feature (see Extended Data Fig. 1). Note all our measurements have been performed within the time interval where \( d(t) \) behavior is linear. Although the resistance of the hydrated membrane is high, for an applied voltage of 10 V, a small current \( (I) \sim 10 \text{ \textmu}A \) passes across the membrane. The observation that \( R \) changes with \( d \) motivated our study of IV characteristics of these membranes at different hydration levels \( (d) \) (see Methods for electrical measurement). The IV curves for different \( d \) in Fig. 1b are non-linear. The shape of the nonlinear IV curves are similar to those studied in molecular junctions\textsuperscript{27,28}. A noteworthy feature in Fig. 1b, is the hysteresis in IV's, viz., slightly different IV paths are traced while changing \( V \) (d.c.) from \(-40 \text{ V} \) to \(+40 \text{ V} \) followed by \(+40 \text{ V} \) to \(-40 \text{ V} \). In Fig. 1b we observe that the mean slope (which is the inverse of the average membrane resistance \( <R^{-1}> \)) of the IV curve reduces with decreasing \( d \), suggesting that the average membrane resistance increases with decreasing water concentration. In Fig. 1a as the variation of \( R \) spans a wide range, hence it appears to be almost constant at \( \sim 5 \text{ M}\Omega \) for variations in \( d > d_{c} \). In Fig. 1c we show the \( R(d) \) measured over a narrower range of \( d \) values, which is akin to viewing the \( R(d) \) behavior in Fig. 1a, on an expanded scale. Here we see the electrical resistance \( R \) is sensitive to relatively small change in the membrane's water concentration, in fact, \( R(d) \) becomes minimum at intermediate values of \( d \). The hysteresis in Fig. 1b is explored further in Fig. 1d by reploting the IV data in Fig. 1b as a dimensionless variable, \( IV^{-1} \), \( <R> \) versus \( V \) (see Methods for scaling analysis details). We found that all the hysteretic IV curves in Fig. 1b at different water concentrations collapsed onto a single master curve which is independent of the variation in \( d \). We have shown that the value of resistance of the membrane is sensitive to variation in \( d \), therefore normalizing the IV data of Fig. 1b using \( <R> \) value results in the implicit dependence on \( d \) being removed and hence all the curves get scaled. The scaling suggests that the shape of hysteresis is unchanged with \( d \) variation, which means the mechanism of charge transport remains unmodified as the water concentration of the membrane changes. The hysteresis in IV also indicates that electric charges flow along trajectories with different resistance as the voltage applied across the membrane is increased or decreased. Due to the complexity of these biological systems, one may expect that their properties are subjected to large variability. Hence, one may ask which aspects of Fig. 1 represents transport properties common to hydrated natural membranes? To answer this question, in the Extended Data Fig. 2 we show the hysteretic nature of the IV curves measured in different samples of hydrated SCM\textsubscript{BMW}.

Transport in biological membranes also depends on the specific chemical nature of the fibers. A reason we study materials with chemically distinct macromolecules, like, silk cocoon membranes, human hair, corn silk and jute, is to check if the charge transport properties differ with the chemical composition of these materials. We have confirmed in the Extended Data Figs 2 and 3, that the nonlinear, hysteretic nature and the scaling properties of IV (similar to Fig. 1) are seen in different hydrated membranes of natural fibers which are chemically distinct as well as in different samples of the same material.

**Temperature dependence of bulk electrical conductivity of these natural membranes.** We explore below the nature of charge transport mechanism through the temperature dependence of electrical conductivity of different natural fibrous membranes and show the common features of this transport mechanism which is shared across different class of hydrated membranes. The linear behavior in Fig. 2a shows the measured
Electrical conductivity of different hydrated membranes obeys the Arrhenius like thermally activated behavior of the form, 
\[ \sigma \propto \exp\left(-\frac{<U>}{k_B T}\right) \], where \(<U>\) is mean energy scale and \(k_B\) is the Boltzmann constant.

The \(<U>\) represents the energy barrier which an electron in the membrane must overcome to conduct current across the membrane in the presence of a biasing electric field (see a schematic of the mechanistic view of thermally activated transport shown in Fig. 2b). The slope of the linear fitted line (solid lines) in Fig. 2a gives an estimate of \(<U>\) ~0.22 eV for both hydrated silk cocoon membrane and human hair and about 0.12 eV for corn silk. Interestingly the jute fiber which is a pure carbohydrate polymer has a low \(<U>\) value i.e., 0.11 eV and it also has the highest electrical conductivity (\(\sigma\)) amongst the natural fibers we have studied. We see that these membranes composed of natural fibers with distinct chemical composition, exhibit similar thermally activated transport behavior.

Figure 1. Electrical transport study of hydrated SCMBMW. For electrical transport measurement, we have used SCMBMW (consisting of natural fibers obtained from the Bombyx mori silk cocoon). The membrane has dimensions of 65 mm (length) \(\times\) 20 mm (width) \(\times\) 0.5 mm (thickness). For all our measurements, electrical contacts are made directly on the membrane at its two ends. The distance between the contacts is approximately 61 mm in present case. Since the resistance of these fibers is very high, the electrical transport measurements are of two probe type. All electrical measurements are done at room temperature. In all our figures, the error bars are smaller than the size of the data symbols. (a) Shows the dependence of resistance (\(R\)) (left axis) on the water concentration (\(d\)) (bottom axis) which varies with time (\(t\)) (top axis). Notably, there is a rapid change in the resistance as the water concentration in SCMBMW falls below a critical value \(d_c = 0.1 \text{ mg mm}^{-3}\). The inset of (a) shows four different runs, 1 (orange), 2 (olive), 3 (violet) and 4 (black) depicting the linear behavior of water concentration as a function of time in SCMBMW with a slope (cyan color) of \(m = -(5.28 \pm 0.04) \times 10^{-5} \text{ mg mm}^{-3} \text{ sec}^{-1}\). (b) The inset is a schematic of the two probe electrical contact on hydrated SCMBMW. The main panel shows a hysteresis in the \(IV\) characteristics of the SCMBMW measured with different water concentration (the \(d\) values are shown). (c) Shows the resistance (\(R\)) measured as a function of time (see bottom axis) or, by using the linear \(d(t)\) relation, as a function of \(d\) (top axis). Here \(R\) is measured with 10 volts applied across the contacts on the SCMBMW. In Fig. 1c we begin the \(R(t)\) measurement with a fresh SCMBMW with an initial water concentration of 0.3 mg mm\(^{-3}\) and then it is allowed to dry naturally. For clarity in seeing the \(R(d)\) behavior, the measurement is performed over a relatively shorter range of variation in \(d\) (compared to that in Fig. 1a). We see that \(R\) is sensitive to the membrane water concentration and seems to show minima at intermediate water concentration. (d) The \(IV\) data in Fig. 1b is normalized and replotted as \(IV^{-1}\). \(<R>\) vs. \(V\). The figure shows that all the different \(IV\) curves of (b) scale onto a single master curve. The scaling property of these curves implies that the \(IV\) hysteresis changes monotonically with water concentrations (\(d\) of the membrane and the shape of the hysteresis also remains unchanged at different values of \(d\).
electrical transport properties and similar IV characteristics (recall Fig. 1, Extended Data Figs 2 & 3 and Fig. 2). As the order of magnitude of \(<U>\) is similar across these different membranes, it suggests that the charge transport mechanism is not dependent on the specific chemical nature of the molecular scaffolds present in these hydrated membranes, viz., the scaffolds of Protein–H2O polymer (present in silk cocoon and human hair), Carbohydrates–H2O polymer (present in jute) and Protein/Carbohydrate hybrid–H2O polymer (present in corn silk) complexes. Our results show that presence of water in these molecular scaffolds is necessary for ensuring charge transport in these membranes. Figures 1 and 2 suggest the interesting possibility of controlling the electrical conductivity of bio-inspired native electrically active materials through the choice of natural membrane type, temperature and hydration levels.

Local electrical characterization of biological membrane (SCM BMW) using STM. The bulk electrical transport studies show the presence of potential wells, which trap the charges in the hydrated biological membranes (see Fig. 2a). We now explore the microscopic local electric potentials present in these hydrated fibers which trap these charges. For this purpose, we use the scanning tunneling microscope (STM) to apply a local electric field using a bias voltage \((V_b)\) between an STM tip and the surface of the material being studied (hydrated SCM BMW membrane in our case). Information about the local electrical properties of the material is revealed by measuring the electron tunneling current established between the atomically sharp STM tip and the surface of a hydrated silk cocoon membrane (SCM BMW) as a function of \(V_b\) (see Methods). In recent times, STM based tunneling current IV measurements have become popular to probe molecular electronic levels of molecular break junctions. In our measurements instead of fabricating nanoscale junction between a STM tip and a molecule, we study the tunneling IV characteristics of the native hydrated membrane. We would also like to mention that our measurements are unlike earlier STM studies, which have been conducted on de-hydrated and usually denatured molecular junctions. Figure 3a shows the STM tunneling current increases from negative values \((-10\) nA) at a negative voltage bias. At the characteristic applied bias \(V_b = -1.25\) V (see the arrow in Fig. 3a)
Identifying the nature of charge carrier in hydrated biological membranes using EPR. In bulk electrical transport investigations, the conventional Hall measurement is used to determine the nature of the charge carriers present in a material. In Hall measurements, a transverse (Hall) voltage appears across a material in a direction which is perpendicular to the plane of the current and an applied magnetic field. However, in our hydrated natural membranes it is difficult to make a transverse Hall voltage contacts. Furthermore, in these membranes due to the presence of a complex crisscrossing network of fibers, the conducting paths of charges are unconventional and hence a Hall measurement would be difficult to interpret. Therefore, to determine the nature of the charge carriers, instead of using transverse Hall measurements, we use the EPR measurement. Figure 4 shows the first derivative of EPR absorption spectra (see Methods) of SCMBMW and human hair membranes in dry and hydrated state (see Extended Data Fig. 3 for jute fiber). The derivative of the EPR absorption spectra of dry and hydrated samples of SCMBMW and human hair in Fig. 4, a and b show that with hydration the center of the EPR absorption spectrum (vertical arrow, \( w_c \)) does not shift, although there is a significant change in the width

**Figure 3.** STM study of SCMBMW. STM study is done on a piece of hydrated SCMBMW with dimension: 9 mm (length) \( \times \) 9 mm (width) \( \times \) 0.5 mm (thickness). (a) Here we have chosen the symbol size to be comparable to the size of the error bars. Tunneling current (I) between an atomically sharp STM tip and the surface of a hydrated SCMBMW membrane is measured as a function of bias voltage (\( V_b \)). The figure adjoining (a) marked by a curved downward pointing arrow shows the schematic of our STM measurement setup, where the tunneling current I is measured between an atomically sharp STM tip and the surface of a hydrated membrane (SCMBMW) placed on an electrically conducting base. The bias voltage is applied between the STM tip and the conducting substrate. A tunneling current is established when the Fermi level of the metallic STM tip coincides with available molecular orbital energy levels in the hydrated membrane (see Extended Data Fig. 4). The tunneling current I vs \( V_b \) is measured for SCMBMW at different water concentration \( d \) (the \( d \) values are shown in the plot). All STM measurements were performed at room temperature. Note I is zero when \( d \) is below the critical value \( (d_c \approx 0.1 \text{mg mm}^{-2}) \). At different values of \( d \), when \( V_b \) is changed from \(-1.25 \text{V} \) to positive values, the tunneling current I increases from \(-10 \text{nA} \) to \(+10 \text{nA} \) by crossing zero at \( V_b = V_p = -1.25 \text{V} \) (shown with arrow). The \( V_p \) reduces to zero as \( d \) decreases. The change in the sign of the tunneling current I at negative value of \( V_b \) we see in (a) is unusual. Inset of (a) shows \( dI/dV \) vs \( V_b \) for different \( d \). See Extended Data Fig. 4 for an explanation of the density of states in the medium. (b–e) Schematics to understand the behavior of sign switching of the STM tunneling current at negative bias value \( V_p \). We consider the membrane develops an internal electric field \( E_{\text{int}} \) when subjected to a bias voltage. The \( E_{\text{int}} \) is assumed to be uniform. When \( V_b < V_p \) and \( V_p < 0 \) (see b), the net electric field \( E_{\text{net}} = E_{\text{app}} + E_{\text{int}} \) is in the direction of \( E_{\text{app}} \) as it is larger than \( E_{\text{int}} \). Near \( V_b = V_p = -1.25 \text{V} \) (Fig. c), the \( E_{\text{app}} \) becomes equal in magnitude and oppositely directed to \( E_{\text{int}} \) (therefore \( E_{\text{net}} \rightarrow 0 \) and consequently the tunneling current measured in the STM approaches zero (arrow position in a). As \( V_b > V_p = -1.25 \text{V} \), \( E_{\text{app}} \) becomes positive and remains positive with further increase in \( V_b \) (d and e) and hence the I changes sign from negative to positive. In these schematics, the orange line (base) represents the contact of SCMBMW with the stub of STM sample holder where we are applying \( V_b \) and tip represents STM tip (in all Fig. from b–e).
of the spectrum (\(\Delta w\)). The hydrated samples of these fibers (SCMBMW, hair and jute) show a narrower spectrum and the EPR intensity also reduces when compared to dry fibers spectrum.

Similar line width narrowing with hydration in silk cocoon membranes has been seen in NMR studies (see figure related to NMR data in ref.1). The narrowing down of the line width is attributed to the hydration induced enhancement in the mobility of molecular components in the macromolecules of silk fibroin, as deduced from neutron scattering experiments34. For both SCMBMW and human hair, \(w_0\) is located between 3.1 to 3.2 kOe. The location of \(w_0\) is comparable with earlier EPR data on SCMBMW37. From \(w_0\), the calculated gyromagnetic ratio (\(g\)) is between 2.008 to 2.010 (see Extended Data, Table 1) implying that the charge carriers are electrons and not protons (the absolute value of the g for electron is ~2.0023, while for protons it is ~5.868) in all these natural fibers. As the measured gyromagnetic ratio represents an electron charge instead of a proton, we exclude the possibility that, charge transport in these hydrated membranes is via protonic, Zundell ion ((H5O2)+) complexes formed by reconfiguring and distorting H-bonds in H2O molecular clusters8,11. The charge carriers in these systems are fundamentally electronic in nature. In Fig. 4c we show that the EPR spectra of a frozen hydrated SCMBMW measured at 170 K (−103 °C, orange curve). Note that its EPR line width (\(\Delta w\)) at 170 K is identical to that of a dry SCMBMW at room temperature. We will discuss these results in the discussion section.

**Dielectric properties of hydrated SCMBMW.** Our STM measurements (in Fig. 3) showed the presence of local internal electric field in the hydrated SCMBMW, whose value depends upon the water concentration within the membrane. The presence of an internal electric field suggests dielectric property of the hydrated silk cocoon membrane. To investigate the dielectric properties of the hydrated SCMBMW membranes, we measure the phase difference between two sinusoidal electrical signals, \(V(t) = 10\sin(\omega t)\) (volt), with \(\omega = 2\pi(500\text{ Hz})\) sent down to two shielded BNC cables (labeled 1 and 2 in the schematic of Fig. 5a). The signals from cables 1 and 2 are measured on the channels, Ch1 and Ch2 of a digital storage oscilloscope (DSO), respectively with a hydrated SCMBMW membrane placed in the path of cable 2 (see the inset of Fig. 5a). Figure 5a shows the Lissajous curves obtained by plotting \(V_{\text{Ch}1}(t)\) (Voltage from cable 2) versus \(V_{\text{Ch}2}(t)\) (Voltage from cable 1, without SCMBMW in its path). Note that, when the cable 2 is electrically continuous (i.e., cable 2 is directly connected to the DSO without the SCMBMW placed in the path of cable 2) then \(V_{\text{Ch}2}(t) = V_{\text{Ch}1}(t) = 10\sin(\omega t)\) (volt), and we observe a straight line with a 45° slope (dark yellow data, labeled as wire) in Fig. 5a. When the SCMBMW membrane is completely dry, viz., d < 0.01 mg.mm\(^{-3}\), as \(V_{\text{Ch}2}(t) \sim 0\), therefore we obtain the vertical (red) line in Fig. 5a. At intermediate levels of water content (d) in the SCMBMW, we obtain elliptical shaped Lissajous curves, as the membrane introduces a phase difference (\(\theta\), a fitting parameter) between the electrical paths through cables 1 and 2. These Lissajous curves are fitted using \(V_{\text{Ch}2}(t) = a_1\sin(\omega t)\), where \(a_1 = 10\) V and \(V_{\text{Ch}2}(t) = a_2\sin(\omega t + \theta)\) where \(a_2\) varies from 8.5 V to 5.5 V as d of the hydrated SCMBMW changes from 0.38 to 0.12 mg.mm\(^{-3}\) (see orange fitted curve through violet data points in Fig. 5a). The \(\theta\) value determined from this fitting, is plotted as a function of d in Fig. 5b (see right axis).

For the hydrated SCMBMW membrane we model the behavior of the phase difference \(\theta\) w.r.t. \(\omega\) as an electrical circuit with bulk equivalent resistance (\(R\)) - capacitance (\(C\)), where the phase difference at \(\omega = \theta = \tan^{-1}\left(\frac{a_1}{a_2}\right)\).

Using this formula and the \(\theta(d)\) behavior from Fig. 1a, the \(\theta(d)\) plot of Fig. 5b is converted into \(C(d)\) behavior (see left axis in Fig. 5b). The capacitance of the hydrated SCMBMW is found to remain constant at about 0.8 nF for a water content \(d > 0.3\) mg.mm\(^{-3}\). As the membrane dries, we see that \(C\) drops rapidly down to 0.4 nF as \(d \rightarrow 0.1\) mg.mm\(^{-3}\). To determine the \(C\) independently, we construct a parallel plate capacitor by placing a piece of hydrated SCMBMW (\(d \approx 0.2\) mg.mm\(^{-3}\)) between the two Cu plates of size 1 cm x 1 cm (see the actual picture of the device in Extended Data Fig. 6). By using the Lissajous curve method described above, we determine a capacitance of hydrated SCMBMW ~3 × 10\(^{-10}\) F (which is consistent with the values in Fig. 5b). Using the expression for
suggests a high dielectric susceptibility, \( \varepsilon \) is the dielectric constant, and for different water concentration \( d \) (values are shown in the plot). Depending on \( d \), the shape of Lissajous curve changes. The inset shows the schematic of the experimental set up. (b) Figure shows dependence of phase difference \( \theta \), between the signals from channel 1 and channel 2, on water concentration \( d \). The behavior of capacitance, \( C \) (left axis) versus \( d \) is also shown. One observes the \( C \) of the membrane drops as the water content \( d < 0.1 \text{ mg mm}^{-3} \).

**Figure 5.** Dielectric property of SCMBMW. Measurement is done on a piece of SCMBMW of the following size: 48 mm (length) \( \times \) 20 mm (width) \( \times \) 0.5 mm (thickness) using Yokogawa DL 9000 series DSO. Electrical contacts are made at the two ends of the sample (distance between contacts \( \approx 46 \text{ mm} \)). (a) In this figure we have chosen the symbol size comparable to the size of the error bars. Lissajous curves are shown (violet, dark cyan and black color data) for different water concentration \( d \). The effective dielectric constant, \( \varepsilon \), of the hydrated SCMBMW is almost double than that of free water which is about 80.8. The high \( \varepsilon \) suggests a high dielectric susceptibility, \( \chi = \varepsilon - 1 = -169 \), implies that the SCMBMW membrane with hydration, gets significantly polarized, electrically. This internal polarization of the membrane is responsible for the local electric fields measured in STM.

**Electric field sensing properties of SCMBMW.** As seen above, due to the relatively high dielectric constant of the SCMBMW, it is likely to make these membranes sensitive to electric fields present in the environment around the membrane. We record the voltage generated across a SCMBMW which is placed at different distances \( r \) from the surface of a shielded copper wire carrying an AC current 0.5 (r.m.s) at 50 Hz. We observe in Fig. 6a that in this non-contact configuration, the hydrated silk cocoon membrane shows a time varying voltage signal pick up (\( V_{pk} \)) across the voltage contacts on the membrane, whenever an AC current flows through the wire (kept at a distance \( r \)). The \( V_{pk} \) signal has a frequency of 50 Hz and the pickup signal strength decreases as the distance \( r \) between the hydrated silk cocoon membrane and the wire increases. We find that up to a distance of 50 cm from the wire surface, the hydrated SCMBMW is capable of picking the electric field modulations present in the environment due to an AC current carrying wire. Note that, we ensured that during these measurements no other current carrying wire was present in a radius of 1 meter around the hydrated membrane (see Supplementary Material Video 1).

In Fig. 6b, we show that SCMBMW sensing property is dependent on water concentration. We keep the SCMBMW at fixed distance from an AC current (50 Hz) carrying wire. We see that the pickup signal strength decreases with \( d \) (all the while \( r \) is fixed). At low water concentration of \( d < 0.1 \text{ mg mm}^{-3} \), the membrane is not sensitive to pick up the 50 Hz signal. In a separate measurement (see Extended Data Fig. 7) we find that beyond \( ~1 \text{ kHz} \), the voltage picked up by the silk cocoon membrane drops.

The property of the hydrated silk cocoon membranes to pick up low frequency electric field modulations from the environment makes it a lucrative option to explore if it is sensitive to bio-rhythmic signals when placed in contact with a human skin.

We keep a hydrated SCMBMW membrane (\( d = 0.52 \text{ mg mm}^{-3} \)) in direct contact with a live human skin and the voltage time series (\( V(t) \)) developed across the two ends of the membrane are recorded on a digital storage oscilloscope (DSO) (see schematic in Fig. 7a). Figure 7a shows how the \( V(t) \) gets modified when the hydrated SCMBMW comes in contact with electric potentials present on the human skin (see \( V(t) \) portion in Fig. 7a marked 'touch'). By taking a Fast Fourier Transform (FFT) of the portion of the \( V(t) \) in Fig. 7a labeled air, we observe that the spectrum shows a distinct peak only around 50 Hz in Fig. 7e, which is the signal of line frequency picked up from the environment (also see Extended Data Fig. 8). The green curve in Fig. 7b is obtained by applying a 49 to 51 Hz band block filter on \( V(t) \) of Fig. 7a to filter out the line frequency and the red curve in Fig. 7c is obtained by further applying a 3 Hz low pass filter on the filtered green curve (filtering is done only on the portion of the signal present inside the square region in Fig. 7b). Figure 7d is the FFT of the \( V(t) \) spectrum in Fig. 7b, and it shows peaks bunched near 22 to 28 Hz and peaks below 3 Hz (Supplementary Material Video 2). We were especially interested in the structure present in the FFT spectrum below 3 Hz, as such low frequency fluctuations are not produced by conventional electronic source (note the absence of 3 Hz and 22 to 28 Hz peak in the FFT spectrum in Fig. 7e when the hydrated SCMBMW...
membrane is placed only in ambient air and not touching any human skin). A possible source of these low frequency signals could be rhythmic processes in the human body like heart beats which are in frequency range of 1 to 2 Hz.

**Discussion**

Our bulk transport measurements have shown that the electrical conductivity of some natural membranes, is poor in their dry state, and it improves with hydration. Local STM measurements show, the tunneling current switches sign (see Fig. 3) at a negative bias voltage $V_p$ instead of zero, signifying the presence of an internal electric field. We estimate the magnitude of this internal electric field to be $E_{int} = \frac{V_p}{d} \sim 2.5 \times 10^4$ V/m, where $s = 0.5$ mm is the membrane thickness (see schematics Fig. 3b–e). The $E_{int}$ represents the hydrated membrane is significantly electrically polarized. This feature is consistent with our measurement of high dielectric susceptibility $\varepsilon_{dry} = -169$ associated with these hydrated membranes. At $V_p = V_{peak}$ the internal electric field inside the hydrated membrane gets compensated by the applied field (Fig. 3c) and thus the tunneling $I \rightarrow 0$ at $V_p = V_{peak}$. This STM measurement shows presence of enhanced density of states (larger $d/dV$) around $V_p$. We observe $V_p$ decreases with $d$ (Fig. 3a) which suggests the internal electric field ($E_{int}$) in the membrane decreases with water content. Also note that with shifting $V_p$, the peak density of states also shifts with $d$ (Fig. 3a inception). When the water content of the SCMBMW is well below 0.1 mg.mm$^{-3}$ then we do not find any evidence of $V_p$ and $d/dV \rightarrow 0$ (i.e., there are no available electronic states for electrons to tunnel into). These results suggest that water controls the generation of local electric fields through polarization produced inside the membrane. This electric field is important for charge conduction in these membranes as we see there are available density of states at $V_p$, which changes with the water content in the membrane.

Through EPR measurements, we identify electrons as the basic charge carrying entity in these natural membranes. These studies also showed an unusual variation in the EPR line width with hydration and temperature. To investigate this issue further, we recall the behavior of the EPR line width (Fig. 4). We found that the EPR line width narrows down with increasing water concentration in the membrane. Also with lowering of temperature as the water in the membrane freezes, the EPR line width broadens and becomes equal to that of the dry membrane at room temperature. The EPR line width changes with the coupling strength of the electron with its local environment. The coupling of the electron to its environment is via spin - spin interaction or via Coulombic interactions. If electric charges in the hydrated membrane were completely free electrons, then one would not have observed changes in the EPR line width. A broadening of the EPR line width when the fibers are dry ($d < 0.1$ mg.mm$^{-3}$), suggests that electrons are strongly coupled to the complex chemical environment in these natural fibers. The narrowing down of the EPR line width with hydration suggests the electron begins to decouple from its complex charged local chemical environment in the presence of water.

From these EPR results it appears that free electron transport is absent in these hydrated, dielectric membrane of the natural fibers. We argue for an alternative mechanism of electron transport in these hydrated fibers: From Fig. 1a recall the sharp increase in resistance below a critical water concentration $d_c = 0.1$ mg.mm$^{-3}$. A $d_c = 0.1$ mg.mm$^{-3}$ corresponds to a density (n) of $3.35 \times 10^{22}$ water molecules.cm$^{-3}$. Using $n$, we make a crude estimate that at $d_c$, the average spacing between water molecules on the molecular scaffolds in the membrane is $\xi = n^{1/3} \sim 6.68$ Å. It turns out that this estimate of $\xi$ is similar to the repeat distance of 7 Å for the R-group in the beta-keratin protein chain of silk cocoon$^{25}$. Based on the correlation that resistance diverges for, $d < d_c$, and $d_c$ is associated with the average water molecule separation becoming of the order of a characteristic length scale on the molecular scaffold of the silk cocoon, suggests that water molecules decorate the molecular scaffold of a

**Figure 6.** Electric field signal pick up by hydrated SCMBMW. For this pick up measurement we have taken a sample of following size: 48 mm (length) × 20 mm (width) × 0.5 mm (thickness). For measurement we have used Yokogawa DL 9000 series DSO. Electrical contacts are made at the two ends of the sample (distance between contacts $\approx 46$ mm). (a) Figure shows the 50 Hz AC signal picked up by hydrated SCMBMW kept at different distances. Inset shows the schematic of experimental set-up. (b) Figure shows the dependence of water concentration on the sensing capability of hydrated SCMBMW. Here we see that when $d < 0.1$ mg.mm$^{-3}$ SCMBMW is not able to pick up the AC signal kept at distance of 2 cm from the wire. Note that in both the figures, the typical level of fluctuation in the data is of the order of 2 mV.

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natural fiber and they affect the electrical conductivity of the fibers. This presence of water molecules decorating the Protein scaffold makes the water-Protein scaffold complex highly susceptible to electric polarization, as the polarizable water molecules can reorient and distort in an electric field. In fact earlier studies on enhancement in dielectric constant of hydrated Proteins have been attributed to the formation of Protein-water complex. On the molecular scaffold decorated by water molecules, an electron traversing along the scaffold, electrically polarizes the hydrated membrane by deforming the water molecules in its local environment (see schematic in Fig. 8). The above feature of the hydrated natural fibers developing additional degree of freedom due to the ability of water molecules to distort, is consistent with observations of increased molecular mobility seen in Neutron scattering experiments, which are considered to narrow down NMR line widths in hydrated silk fiber.

The electric field of an approaching electron polarizes (see Fig. 8) the water molecules momentarily disturbing the positively charged hydrogen (shaded grey) end distorting towards the electron. This leads to the formation of an intermediate composite state, viz., a Polaron complex (see schematic in Fig. 8). The Polaron complex is an electron surrounded by cloud of positive charge fluctuations generated by the distorting hydrogen ends of water molecules. The creation of the positive charge cloud in turn attracts the electrons in its vicinity and in turn facilitates charge conduction along the molecular scaffold. Note that Polaron mediated charge transport mechanism is a well-known mechanism in some condensed matter systems, like, the high resistance dielectric organic polymers. A Polaron requires distortion of the water molecules, therefore the characteristic energy of the Polaron can be considered to be $\hbar \omega$, where $\omega$ is a characteristic vibration frequency in this system. The energy of 0.2 eV (see typical $< U >$ values in Fig. 2a) we propose is associated with the energy of a Polaron. This energy scale of 0.2 eV is in the same range of as the intramolecular vibrational transitions in the mid IR regime of water molecule. This observation supports our view that charge transport along the molecular scaffold occurs by exciting
vibrational modes (distorting water molecules) in the water molecules, whose energies are in the range of 0.2 eV. It may also be mentioned in passing that 0.2 eV Polars have been reported in other non-biological systems.41

We now show that the reduced effective charge on the Polaron can also be estimated from our experimental results in the following way: Recall our STM measurements showed the presence of a low internal field inside the hydrated membrane which corresponds to a characteristic potential difference of magnitude, ∆Vp, ~ 1.25 V (see Fig. 3). Inside the hydrated membrane, trapping of charges by the local electric field is related to the activation barrier, U0, ~ 0.2 eV, determined from the temperature-dependent bulk conductivity measurements of Fig. 2. For a Polaron with net effective charge q, in order for it to hop it does work against the internal electric field, i.e., it does a work qVp, with U0 ~ qVp. Using the values of U0 and ∆Vp, we estimate the Polaron has a net effective charge q ~ 0.16e, where e is the charge of a free electron. Thus shielding of the free electrons by the positive charge cloud in the Polaronic state, significantly reduces its net charge compared to that of a free electron. An indirect evidence of reduction of the effective electrical charge on the charge-carrying entity in a hydrated membrane is hydration induced narrowing down of the EPR line width. A Polaron with lower effective charge is electrostatically more weakly coupled to the local charged environment on the molecular scaffold. Hence a narrower EPR line width results (Fig. 4a–c) when the membrane is hydrated. Recall in Fig. 4c we had shown that the EPR spectra of a frozen hydrated SCM BMW measured at 170 K (−103 °C, orange curve). Note that its EPR line width (Δω) at 170 K is identical to that of a dry SCM at room temperature. At low temperature the water molecules freeze thereby reducing their degree of freedom on the molecular scaffold and hence, they cannot distort easily to create a Polaron when an electron approaches. Therefore, as a Polaron cannot form, hence ineffective shielding of the electron’s charge at low temperature enhances the coupling of the electron to its local charged chemical environment which leads to a broadening of the EPR spectra in the frozen state back to a level which is comparable to that in the dry state of the membrane (Fig. 4c). Apart from temperature, enhanced electrostatic repulsion between water molecules at higher water molecule density also hampers their ability to distort or reorient water molecules. Hence, high concentration of water molecules in the membrane would also inhibit formation of a Polaron (as distortions of water molecules are not favored) and hence R increases at high water concentrations (d) (see Fig. 1c). The formation of Polaron leads to enhanced density of states found in Fig. 3a inset for the hydrated natural fiber (also see schematic in Extended Data Fig. 4). Due to the electrically polarizable nature of the hydrated medium of the natural fiber, application of V affects the dielectric medium and also its conductivity. Therefore, it is not surprising to observe the nonlinear nature of IV as shown in Fig. 1b.

Furthermore, an outcome of our study is that these hydrated electrically polarizable membranes are potentially useful as sensors to detect electrical fields in the environment or in live human tissues (see Figs 6 and 7). From our measurements we know that the typical membrane resistance is ~10 mΩ with a capacitance ~0.1 nF. This gives a typical R/C time constant ~1 msec (or (RC)−1 ~1 kHz) for these membranes. Hence the typical band-width of these bio-membrane sensors is from low frequencies up to 1 kHz which is consistent with our measured cut off frequency (see Extended Data Fig. 7). This gives a unique advantage to these sensors, as they detect electric field modulations in the low frequency regime, whereas other conventional sensors in this regime have low sensitivity due to 1/f noise considerations.

Conclusions
In conclusion, we have uncovered the important effect that water plays in not only producing dielectric polarizability in natural fibers but also helps in mediating electron transport across the fibers. Due to the enhancement
in their dielectric polarizability, the hydrated natural fibers develop ability to sense electric fields. This property makes these hydrated natural membranes potentially useful for application as sensors for bio-rhythmic processes which are associated with low frequency electric field modulation. Through our present study performed on at least ten samples of silk cocon membranes, jute and corn silk fibers, we have uncovered features of electron transport in these hydrated bio-polymers. While our STM measurements with a sensitivity to detect electric fields of about 0.5 V/m, have revealed the presence of local electric fields (~10³ V/m) inside the hydrated silk cocon membrane, its spatial distribution and fluctuation in time are still undetermined. These properties need to be mapped in detail on the hydrated silk protein scaffold in its native state in order to establish the details of the Polaronic mechanism. We also need to have high resolution imaging of the molecular fabric of the hydrated scaffold at different temperatures. Our method to measure the dielectric property of these membranes, is viable for measurement of dielectric properties as a function of water content and varying temperature. However, the systems measurement sensitivity is limited to about 0.1 nF capacitance, and the technique cannot determine changes in the dielectric properties of the material at a local level or if the quantity of the material being probed is very small. In the future we believe a better understanding of the charge transport mechanism in natural fibers will have potential ramifications for understanding the behavior of electrically active biological tissue.

Methods

Electrical Transport Measurement. Electrical measurements are performed at room temperature and pressure using Keithley 2400 source meter. The membrane of Bombyx mori silk cocon (SCM BMW) is first cut into small rectangular pieces (size of the membrane is mentioned in the figure captions in main manuscript) and then dipped in deionized water to hydrate the membrane (the purity of distilled water is 2 ppm i.e., there are 2 mg of dissolved solids in 1 kg of water). Electrical contacts are made at the two ends of the membrane. Contacts are made with aluminum (Al) and copper (Cu) wire at two ends (we have also tried with Cu-Cu contacts and electrical characteristics were same in both cases). See section on Image of samples with electrical contacts in Supplementary Information.

For making hair, jute and corn-silk membranes we have used a collection of hair, jute and corn-silk fibers and electrical contacts are made at the two ends of these membranes. Contacts are made with aluminum (Al) and copper (Cu) wire at two ends. Measurements have been done with four probe and two probe. No difference was found between the four probe and two probe measurements as the resistance of the medium is very high. All transport measurements reported in the paper are two probe.

We measure resistance (R = V/I) of these membranes as a function of its water content (d) by initially hydrating the samples and then allowing them to dry at 300 K. To get the information about water content in the membrane, first, we measure the weight of dry sample. After hydrating the membrane with water we put it on a weighing machine and noted the weight after a fixed interval of time. After subtracting these values from the dry weight we get the water concentration (d) in membrane by determining the weight to the volume ratio.

Scaling of IV plot. To check if the nature of the hysteresis in the IV curve remained unchanged with varying water content, we scaled them in the following way. Each forward branch of IV (viz., we have measured I as V is swept from a −ve to a +ve value and back to −ve value) in Fig. 1b is fitted to a linear form, I = (R< −1) V, where <R> = 1/2 IV curve is divided by the mean value of the slope.

STM (Scanning Tunneling Measurement). The IV measurements of the hydrated silk cocon is carried out in a Quazar Technologies made room temperature scanning tunneling microscopy system (STM) (NanoRev. 4.0). In our measurement we use an atomically sharp tip fabricated from a Pt-Ir alloy (using electrochemical etching process). The hydrated membrane SCMBMW is mounted in the STM on a gold coated metallic stub (sample holder) and the bias voltage, Vb, is applied between the metallic stub and the Pt-Ir tip which is kept at ground potential. When the tip is not touching the membrane and there is a gap of few nm between the STM tip and the surface of the SCMBMW membrane, we observe no appreciable tunnel current. For measuring any appreciable tunneling current we do the following: while maintaining a finite bias voltage Vb between the STM tip and the metallic stub on which the SCMBMW membrane is placed, the STM tip is first lowered towards the surface of the membrane until the tip touches the membrane. At this point a large current is observed. We then begin retracting the tip so that the tip begins leaving the contacted region in the membrane in steps of size 2 Å. During this process at every retracted position, the tip is held stationary and in this position the tunneling current I versus bias voltage Vb, curve is measured. The process of measuring I versus Vb curve is continued at every retracted step of the STM tip, until the tunnel current disappears. We find that the STM tip needs to be retracted by about 20 to 30 Å to reach the zero tunneling current state. I versus Vb curves shown in Fig. 3a have been obtained just before the tunneling current disappears.

EPR (Electron Paramagnetic Resonance Measurement). The unpaired electron of a material and the electronic structure was studied by electron paramagnetic resonance (EPR). Electron has a spin magnetic moment $\mathbf{S} = 1/2$, with magnetic component $m_s = \pm 1/2$. When a paramagnetic material with unpaired electrons is placed in an external magnetic field $B_0$, due to the Zeeman effect degenerate energy levels of the unpaired electrons are split in two energy levels with energy difference $\Delta E = g_s \mu_B B_0$, $g_s$ is the Lande g factor and $\mu_B$ is the Bohr magneton. If we apply an electromagnetic (EM) signal in the presence of a magnetic field, at the resonance condition an absorption peak is observed. This is the central idea of EPR spectroscopy. If $\nu$ is the applied frequency of the EM signal, then at resonance $h \nu = g_s \mu_B B_0$. 
EPR measurement is done using “Bruker EMX EPR Spectrometer”. EPR measurement is carried out at a constant frequency of 9.864 GHz and by varying the magnetic field. Sample is placed in the quartz tube. At first magnetic field is varied over the full range from 0 to 8000 Oe. The precise measurement is done close to the peak region in the absorption spectra. To eliminate any geometric effects affecting the measurements, we have made symmetric bundle of the sample. At first we take the measurement of the sample with the quartz tube. Data is plotted after background (quartz) subtraction.

Online Content. Additional Extended Data display items and Source Data, are available in the online version of the paper; references unique to these sections appear only in the online paper.

References
1. Tulachan, B. et al. Electricity from the Silk Cocoon Membrane. Sci. Rep. 4, 5434 (2014).
2. Tulachan, B., Singh, S. K., Philip, D. & Das, M. Harvesting electricity from human hair. Cosmet Sci 67, 21–36 (2016).
3. Alberts, B. et al. Molecular Biology of the Cell 4th edn (New York: Garland Science; 2002).
4. Daniel, R. M., Finney, J. L. & Stoneham, M. ‘The molecular basis of life: is life possible without water?’ Phil. Trans. R. Soc. Lond. B 359, 1143 (2004).
5. Ball, P. Water: Water - and enduring mystery. Nature 452, 291–292 (2008).
6. Ball, P Water as an Active Constituent in Cell Biology. Chem. Rev. 108, 74–108 (2008).
7. Karakha, L. M. & Kreuzer, J. H. Charge transport along proton wires, Biointerphases 8, 13 (2013) and references therein.
8. Okimoto, N. et al. Cooperative Motions of Protein and Hydration Water Molecules: Molecular Dynamics Study of Scytalone Dehydratase. J. Am. Chem. Soc. 126, 13132–13139 (2004).
9. Olanó, L. R. & Rick, S. W. Hydration free energies and entropies for water in protein interiors. J. Am. Chem. Soc. 126, 7991 (2004).
10. Dwyer, J. J., Gittis, A. G., Karp, D. A., Lattman, E. E. & Spencer, D. S. High apparent dielectric constants in the interior of a protein reflect water penetration. Biophys J. 79, 1610–1620 (2000).
11. Nagle, J. F. & Morowitz, H. J. Molecular mechanisms for proton transport in membranes. Proc. Natl Acad. Sci. USA 75, 298–302 (1978).
12. Pomès, R. & Roux, B. Quantum effects on the structure and energy of a protonated linear chain of hydrogen-bonded water molecules. Chem. Phys. Letters 234, 416–424 (1995).
13. Agnon, N. The Grotthuss mechanism. Chem. Phys. Letters 244, 456–462 (1995).
14. Pomès, R. & Roux, B. Theoretical Study of H+ Translocation along a Model Proton Wire. J. Phys. Chem 100, 2519–2527 (1996).
15. Pomès, R. & Roux, B. Structure and dynamics of a proton wire: a theoretical study of H+ translocation along the single-file water chain in the gramicidin A channel. Biophys J. 71, 19–39 (1996).
16. Meil, H. S., Tuckerman, M. E., Sagnella, D. E. & Klein, M. L. Quantum Nuclear ab initio Molecular Dynamics Study of Water Wires. J. Phys. Chem. B 102, 10446–10458 (1998).
17. Decornez, H., Drukker, K. & Hammes-Schiffer, S. Solvation and Hydrogen-Bonding Effects on Proton Wires. J. Phys. Chem. A 103, 2891–2899 (1999).
18. Nemukhin, A. V., Grigorenko, B. L., Topol, I. A. & Burt, S. K. Quantum chemical simulations of the proton transfer in water wires attached to molecular walls. J. Phys. Chem. B 107, 2958–2965 (2003).
19. Braun-Sand, S., Burykin, A., Chu, Z. T. & Warshel, A. Realistic Simulations of Proton Transport along the Gramicidin Channel: Demonstrating the Importance of Solvation Effects. J. Phys. Chem. B 109, 583–592 (2005).
20. Jaenicke, S. et al. Field-assisted photodesorption of ions from metal and semiconductor surfaces. J. De Physique (Paris) 47-C7, 343–347 (1986).
21. Grotthuss, C. J. T. de. Sur la décomposition de l'eau et des corps qu'elle tient en dissolution à l'aide de l'électricité galvanique. Ann Chim 58, 54–73 (1806).
22. Roy, M. et al. Carbon dioxide gating in silk cocoon. Biointerphases 7, 1–11 (2012).
23. Creighton, T. E. Proteins: Structures and Molecular Properties (W. H. Freeman, 1992).
24. Popescu, C. & Hocker, H. Hair—the most sophisticated biological composite material. Chem Soc Rev 8, 1282–91 (2007).
25. Sirkar, S. C. & Saha, N. N. Hydrated cellulose from Jute Fibre. Nature 156, 839–839 (1946).
26. Sarepoua, E., Tangwongchai, R., Suriharn, B. & Lertrat, K. Influence of variety and harvest maturity on phytochemical content in corn silk. Biophys J. 8, 29214 (2016).
27. Roy, M. et al. Soft magnetic memory of silk cocoon membrane. Sci. Rep. 6, 29214 (2016).
28. Tulachan, B. et al. The role of photo-electric properties of silk cocoon membrane in pupal metamorphosis: A natural solar cell. Sci. Rep. 6, 21915 (2016).
29. Nitzan, A. & Ratner, M. A. Electron transport in molecular wire junctions. Science 300, 1384 (2003).
30. Zimbovskaya, N. A. & Pederson, M. R. Electron transport through molecular junctions. Phys. Rep. 509, 1 (2011).
31. Xu, B. Q. & Tao, N. J. Measurement of Single-Molecule Resistance by Repeated Formation of Molecular Junctions. Science 301, 1221 (2003).
32. Haiss, W. et al. Redox State Dependence of Single Molecule Conductivity. J. Am. Chem. Soc. 125, 15294–15295 (2003).
33. Jung, S. Y., Reddy, P., Majumdar, A. & Segalman, R. A. Interpretation of Stochastic Events in Single Molecule Conductance Measurements. Nano Lett. 6, 2362–2367 (2006).
34. Guo, S., Hihath, J., Díez-Perez, I. & Tao, N. Measurement and Statistical Analysis of Single-Molecule Current-Voltage Characteristics, Transition Voltage Spectroscopy, and Tunneling Barrier Height. J. Am. Chem. Soc. 133, 19189–19197 (2011).
35. Irena, K. et al. Conductivity of natural and modified DNA measured by scanning tunnelling microscopy: The effect of sequence, charge and stacking. Biophysical Chemistry 138, 3–10 (2008).
36. Seydel, T. et al. Increased molecular mobility in humid silk fibers under tensile stress. Phys. Rev. E 83, 016104 (2011).
37. Deng, Y. B., Cai, J. H. & Zhou, P. Naturally Stable Free Radical in the Silk Fibroin and its Structure Environment Studied by EPR and DFT. Spectroscopy Letters 45, 285–295 (2012).
38. Devreeze, J.T. ‘Polaron’ in Encyclopedia of Physics, Vol.2 edited by R. G. Lerner and G. L. Trigg (Wiley-VCH, Weinheim 2005) p.2004.
39. Diasnukes, G. C., McGuire, A., Blankenship, R. & Sauer, K. Electron spin polarization in photosynthesis and the mechanism of electron transfer in photosystem I. Experimental observations. Biophys. J. 21, 239–256 (1978).
40. Aguirre, A. et al. Multifrequency EPR analysis of the positive polaron in 12-doped poly(3-hexylthiophene) and in poly[2-methoxy-5-(3,7-dimethoxyloctyl)-1,4-phenylenevinylene. Phys. Chem. Chem. Phys. 10, 7129–7138 (2008).
41. Schirmer, O. F., Imlau, M., Merschjann, C. & Schoke, B. Electron small polarons and bipolarons in LiNbO3. J. Phys.: Condens. Matter 21, 123201 (2009).

Acknowledgements
SSB thanks the Central EPR facility at IIT Kanpur, funding from DST and IIT Kanpur. SSB thanks Ms. Kamalika Nath for helping in the initial measurements.
Author Contributions
A.K. and A.J. performed all experiments in S.S.B.’s lab at IITK and they helped in writing of paper and doing analysis. A.D. developed the natural fiber based membranes with electrical contacts and conceived the idea of using corn silk fiber for these studies. A.B. conceived the biosensor concept and was involved in scientific discussions on experimental results and helped in writing the manuscript along with S.S.B. D.P. developed the energy harvesting concept using silk and hair biomaterials. K.B. did screening of the different fibers and studying their molecular properties. S.K.S. designed the energy harvesting devices assisted in writing the manuscript. M.D. with A.B. proposed the possibility to the bio-sensors potential of these fibers. M.D. demonstrated the energy harvesting potential of these materials by showing their thermoelectric properties and he was involved in discussions related to the experimental results and contributed to writing the manuscript along with S.S.B. S.S.B. conceptualized the measurements to be performed, as well as planned and guided, the measurements and data analysis of, electrical transport, STM, EPR and dielectric susceptibility. S.S.B. conceptualized time series measurements of the signals on DSO from hydrated SCM_{BSM} when in contact with human skin, in order to discern the structure in signal from random fluctuations. S.S.B. with A.K. and A.J., setup and performed the first measurements of the bio-sensor aspect of the hydrated silk-cocoon for picking up time series of signals from human skin tissue. S.S.B. proposed the interpretations and also wrote the manuscript.

Additional Information
Supplementary information accompanies this paper at https://doi.org/10.1038/s41598-018-20313-4.

Competing Interests: The authors declare no competing interests.

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