Morphological, Structural, and Optical Bandgap Characterization of Extracted ZnO Nanoparticles from Commercial Paste

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1. Introduction

Zinc oxide (ZnO) is a compound of group II–VI. It is a broadband semiconductor oxide with a direct bandgap of approximately 3.37 eV at room temperature [1]. ZnO has a high chemical and mechanical stability. In addition, it is nontoxic and abundantly available in nature. ZnO became a material of interest for different technological applications due to its high optical transmission in the visible light range. Such applications include optical devices, optoelectronics, flat screens, liquid crystal displays, transparent electronics, and thin-film photovoltaic devices [2–4]. With the advance of nanotechnology, considerable research efforts have studied the properties of ZnO in its nano-dimensional scale, most commonly as thin films. ZnO films attract a lot of interest for their applications in optoelectronics and photovoltaics. One of its applications is acting as an electron transport layer (ETL) in third-generation solar cells due to its high electron mobility [5]. Thin layers of nanostructured ZnO can function as ETL in solar cells that include organic [6], dye sensitized [7], and perovskite solar cells [8], respectively. In addition, when obtained as a mesoporous structure, it could have a possible industrial application for conjugated materials [9, 10]. Several deposition techniques are used to grow ZnO films, such as thermal evaporation [11], spray pyrolysis [12], sputtering [13], laser ablation [14], and sol-gel [15] among others. In this work, we extracted the ZnO nanoparticles (NPs) from a commercial paste, in a colloidal form similar to that obtained through the conventional sol-gel chemical process. We obtained ZnO thin films by depositing the extracted ZnO NPs on corning glass substrates by spin-coating. Then, we used different
characterization methods to assess their possible applications in third-generation solar cells. This work opens the door to the recycling and reuse of nanostructured oxides after recovering them from their original products. The recovered oxides can be used in different applications, such as the fabrication of solar cells to produce clean, cheap, and abundant energy.

2. Materials and Methods

3.36 grams of potassium hydroxide (KOH) were dissolved in 200 ml of water, while 30 grams of the commercial ZnO paste (bacitracin zinc 500 [USP’U] in 1 g, polymyxin B sulfate 10000 [USP’U] in 1 g, neomycin sulfate 3.5 mg in 1 g, pramoxide hydrochloride 10 mg in 1 g, Genomma lab USA Inc) in a laboratory beaker were placed on a magnetic stirring hot plate, applying a low temperature. When the paste melted, the temperature rose to 60°C, and the KOH solution was dripped slowly using a Pasteur pipette. Magnetic stirring was applied during the KOH dripping, stirring at 200 rpm, which increased with increasing the volume of the mixture in the beaker until it reached almost 400 rpm. After completing the dripping of the KOH solution, the sample remained under magnetic stirring of 400 rpm at 60°C for 30 minutes and then removed from the magnetic stirring hot plate and placed in a stable place to cool down. During the cooling process, three layers appeared in the beaker as shown in Figure 1: a top layer of the organic materials found in the commercial paste, a layer of ZnO precipitate (PPT) at the bottom, and a colloid of KOH solution along with ZnO NPs in the middle. After twenty-four hours from the previous process, the top layer was separated from the beaker, and a sample was taken from the colloid layer using a pipette. Then, a magnetic stirring of 400 rpm was applied to the colloid and the precipitate in the beaker for 18 hours. After the stirring period, the beaker was left 24 hours to stabilize, and a second sample of the colloid was taken. The two colloid samples were then characterized by the Zetasizer analysis.

Thin films of ZnO were deposited on clean glass substrates of 1 cm² using a mixture from the colloid and the precipitate. The mixture was stirred for 30 min at 400 rpm before the deposition to form a homogeneous suspension. The uniformity of the deposited thin films was improved after various experimental trials, changing the spin-coating parameters without changing the preparation method of the mixture. The films were deposited by a series of successive layers, using dynamic spin-coating with a rotation speed of 3000 rpm, rotation time of 15 seconds, and 5 minutes of preheating between layers at 120°C. After the deposition of all stacked layers, the films were annealed at 550°C for 60 minutes. In this work, we report the results of ZnO thin film deposited in 5 stacked layers. The film was characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), energy dispersive spectroscopy (EDS), and optical absorption spectra measurements.

Malvern Zetasizer Nano ZS DL was used to determine the size range of the NPs extracted from the paste. Figure 2(a) shows the result of the analysis performed on the first colloid sample, taken before the magnetic stirring. Two curves are observed: the first shows NP sizes ranging from 45 nm to 190 nm, which represents 92.4% of the amount of ZnO NPs, and the second shows NP sizes ranging from 190 nm to 825 nm, which represents 7.6% of the amount of ZnO NPs in the colloid. Figure 2(b) shows the result of the Zetasizer analysis performed on the second sample, taken after applying a magnetic stirring of 400 rpm for 18 hours. There is a significant decrease in the volume of the second curve (particle sizes from 190 nm to 825 nm), while the first curve became sharper (particle sizes from 35 nm to 100 nm), representing 97.5% of the amount of ZnO NPs in the colloid. These results demonstrate that the developed extraction process successfully extracted the ZnO NPs from the commercial paste, while the subsequent magnetic stirring homogenized the extracted particles and reduced their dimensions.

3. Results and Discussion

3.1. Zetasizer. The Zetasizer analysis was used to determine the size range of the NPs extracted from the paste. Figure 2(a) shows the result of the analysis performed on the first colloid sample, taken before the magnetic stirring. Two curves are observed: the first shows NP sizes ranging from 45 nm to 190 nm, which represents 92.4% of the amount of ZnO NPs, and the second shows NP sizes ranging from 190 nm to 825 nm, which represents 7.6% of the amount of ZnO NPs in the colloid. Figure 2(b) shows the result of the Zetasizer analysis performed on the second sample, taken after applying a magnetic stirring of 400 rpm for 18 hours.

3.2. SEM. SEM images were used to study the surface morphology of the deposited ZnO films. Figures 3(a) and 3(b) show SEM micrographs of the ZnO film at different spots and magnifications. The images show a mesoporous...
arrangement of nano and submicronic particles. There are variations in the shape, diameter, and crystal size of the particles. The smaller particles represent the NPs in suspension within the colloid as per the Zetasizer analysis, while the larger particles represent the submicronic particles from the precipitate. Particles have different orientations, which indicate the existence of different crystalline planes of the material.

To identify the size range of the precipitate particles, we measured the larger ZnO particles using a high-quality image with a magnification of ×50000, as shown in Figure 4. The particles size varied from less than 100 nm to just over 400 nm.

Figure 2: Zetasizer analysis results (a) before magnetic stirring and (b) after magnetic stirring.

Figure 3: SEM images of ZnO film, showing the morphology at different magnifications: (a) ×5,000 and (b) ×20,000.

3.3. EDS. The EDS analysis identified the characteristic peaks of oxygen and zinc, as shown in Figure 5. No other elements were identified, which indicates the purity of the deposited ZnO. According to the EDS report, the weight percentages and atomic percentages of Zn and O in the film are 76.81; 23.19 and 55.22; 44.78, respectively, which is close to the bulk ZnO weight percentage (80 for Zn and 20 for O) [16]. The EDS analysis did not identify any of the other small peaks shown in Figure 5. They either represent negligible traces of other elements from the paste or noise from the amplifiers and microphonics of the equipment (a typical problem associated with EDS measurements).
3.4. XRD. Figure 6(a) shows the XRD analysis result of the ZnO thin film. The XRD pattern of the thin film shows polycrystalline phases with the following $h, k, l$ Miller indices: (100), (002), (101), (102), (110), (103), (200), (112), and (201). This result coincides with the standard ZnO pattern in the hexagonal wurtzite phase (crystallographic chart JCPDS #36-1451). There is no evidence of remnant materials or impurities as no distant peaks were detected. The sharp diffraction peaks indicate good crystallinity of the NPs in the film. The peaks are slightly shifted to the left in comparison to the standard ZnO peaks, as shown in Figure 6(b) and quantified in Table 1. This shift could be a result of the annealing temperature used to deposit the ZnO film [17, 18], which is considered as an expansion deformation in the crystal lattice of the ZnO particles.

The average crystallite size ($D$) was estimated using the Scherrer formula [19]:

$$D = \frac{K\lambda}{\beta_{hkl} \cos \theta}$$  \hspace{1cm} (1)

where $\lambda$, $\beta$, and $\theta$ are the X-ray wavelength, the full width at half maximum (FWHM) height of the diffraction peak, and Bragg’s diffraction angle, respectively. The average crystallite size calculated from the diffraction peaks was 43.25 nm. The induced strain $\varepsilon$ due to imperfection and distortion of the crystal in each plane was calculated using the following formula [20]:

$$\varepsilon = \frac{\beta_{hkl}}{4 \tan \theta}$$  \hspace{1cm} (2)

The average strain value calculated from the diffraction peaks was $1.954 \times 10^{-3}$. The detailed results are presented in Table 1.

For the hexagonal structure of ZnO, the space between the different planes is related to the lattice parameters of the crystal structure $a$ and $c$ and the Miller indices $h, k, l$ by the following relationship [19]:

$$\frac{1}{d_{(hkl)}^2} = \frac{4}{3} \left( \frac{h^2 + hk + k^2}{a^2} \right) + \frac{4}{3} \left( \frac{h^2 + k^2 + hk}{c^2} \right)$$  \hspace{1cm} (3)

Given that $d = n\lambda/2 \sin \theta$ and with $n = 1$ (first-order approximation),

$$\sin^2 \theta = \frac{\lambda^2}{4a^2} \left[ \frac{4}{3} (h^2 + k^2 + hk) + \left( \frac{a}{c} \right)^2 \right].$$  \hspace{1cm} (4)

The parameter $a$ for the plane (100) is calculated by the following:

$$a = \frac{\lambda}{\sqrt{3} \sin \theta}$$  \hspace{1cm} (5)

The parameter $c$ for the plane (002) is calculated by the following:

$$c = \frac{\lambda}{\sin \theta}$$  \hspace{1cm} (6)

The volume $V$ of the crystal is calculated by the following:

$$V = \left( \frac{\sqrt{3}}{2} \right) (a^2) (c).$$  \hspace{1cm} (7)

The lattice parameter values of the deposited ZnO thin film are slightly higher than those of the standard ZnO, as shown in Table 2. This indicates a small expansive deformation in the crystal lattice of the ZnO and confirms what was indicated by the small shift in the XRD diffraction peaks (Figure 6(b)). This small expansive deformation results in an increase in 0.728% in the volume of the ZnO crystal.

The dislocation density ($\delta$), which represents the number of defects in the sample, is defined as the length of the dislocation lines per unit volume of the crystal and is calculated using the following equation [21]:

$$\delta = \frac{1}{D^2}$$  \hspace{1cm} (8)

where $D$ is the average crystalline size. The calculated dislocation density ($\delta$) is $5.35 \times 10^{-2}$ nm$^{-2}$.

The Zn–O bond length is calculated by the following formula [22]:

$$L = \sqrt{\left( \frac{a^2}{3} + \left( \frac{1}{2} - \frac{u}{2} \right)^2 \right) c^2},$$  \hspace{1cm} (9)

where $a$ and $c$ are the crystalline parameters and $u$ is the positional parameter in the wurtzite structure, a measure of the amount in which each atom moves with respect to the next one along the $c$ axis. The parameter $u$ is determined by the following equation:
3.5. Optical Properties. Figure 7 shows the optical absorbance spectrum of the ZnO film. The band gap energy estimated from the absorption spectrum is ~3.80 eV which is higher than the standard value of ZnO (3.37 eV).

Given the particle size range detected in the Zetasizer analysis (35–100 nm) and the average crystalline size calculated for the deposited ZnO film (43.25 nm), the higher optical bandgap value could be due to quantum confinement, which modifies the bandgap value of semiconductor materials. In general, the structural and optical properties of the sample coincide with what other studies reported for ZnO synthesized by sol-gel and deposited by spin-coating [23–25]. In this work, we recovered ZnO NPs from a commercial product through a process that implies simplicity and low cost.

4. Conclusions

We chemically extracted the ZnO particles from a commercial product (Dr. Bell’s Pomade) and deposited them on glass substrates by spin-coating. The ZnO thin film is
mesoporous, with uniformly arranged nanostructured particles of different shapes. The film is formed by pure hexagonal wurtzite ZnO, with no impurities or remnants of other materials. The average crystalline size of the ZnO particles is ∼43.25 nm. The particles showed a slight volume expansion of 0.728% due to a tensile strain of 1.954 × 10⁻³. All calculated structural parameters are similar to those of the standard ZnO values. The estimated bandgap from the transmittance spectrum is larger than the standard bandgap of ZnO. A quantum confinement effect could explain the larger bandgap. Our work opens the door to the recycling of nanoparticles, an area that has little research to date but great importance and potential for the future [26]. The reported ZnO film has suitable properties for optoelectronic and photovoltaic applications. One of the possible applications is as an ETL in third-generation solar cells.

Data Availability

The data used to support the findings of the study are available from the corresponding author upon request.

Conflicts of Interest

The authors declare that there are no conflicts of interest regarding the publication of this paper.

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