An investigation was carried out to characterize the microstructure and bonding properties of the zirconium-carbon steel explosive clad. The microstructure and the composition of the clad were characterized using optical microscopy and scanning electron microscopy. Bonding properties were inspected by using bending and shearing tests. The examination results indicate that the R60702 and Cr70 plates were joined successfully without visible defects. The interface wave is symmetrical. There is no element diffusion across the interface of the clad plate. There are melt blocks at the interface. Bending and shearing test results indicate that the bonding properties of the clad meet the requirements of the ASTM B898 specification. And after shell rolling, no delamination appeared at the interface. Thus, it indicates that the clad plates have good bonding quality and meet the processing requirement.

1. Introduction

Zirconium (Zr) and its alloy have excellent corrosion resistance in acid and alkali environment. They show more excellent corrosion resistance than stainless steel, titanium alloy, and nickel alloy. Therefore, Zr is widely used in harsh environments, such as dehydration tower of acetic acid project in the chemical industry, synthesis tower in urea project, and reactor in alcohol production [1–7]. However, Zr is difficult to separate hafnium during refining. Therefore, the price of industrial pure zirconium is relatively expensive, which is 4-5 times that of high-grade stainless steel and 2-3 times that of titanium. Therefore, in recent years, zirconium-steel clad materials are used to replace pure Zr in chemical equipment. The processing method of zirconium-steel clad material is usually explosive welding. In this method, the explosive is used as the energy source, so that the atoms between the two metal interfaces are infinitely close, thus forming a firm combination. However, the physical and chemical properties of zirconium and steel are very different, so it is difficult to clad zirconium and steel by explosive welding method. Thus, Ti is used to the interlayer between zirconium and carbon steel to bond them. The microstructure and properties of zirconium-steel clad materials with Ti interlayer have been investigated by more studies [8–15]. Generally, the interface is the weakness of the clad plate, but the zirconium-steel clad with interlayer has two interfaces. Thus, the study on the zirconium-steel clad materials without the interlayer is useful. However, not too much work has been implemented on zirconium-steel clad materials without the interlayer. Therefore, in this paper, the R60702-Gr70 clad material without the interlayer was prepared by explosive welding. The microstructure and mechanical properties of the clad material were characterized. The results show that the clad material meets the requirements of specification and processing.

2. Experimental Procedure

2.1. Materials. The flyer and the base plate are zirconium (ASME SB551 R60702) in solution-annealed condition and carbon steel (ASME SA Gr70) in normalization condition, respectively. The chemical compositions of R60702 and Gr70 are listed in Table 1 and Table 2. The flyer and base
plate were sequentially prepared with dimensions of 3 × 2330 × 6000 mm and 25 × 2330 × 6000 mm, respectively.

### 2.2. Explosive Welding Process

Figure 1 shows the assemble sketch of explosive welding. The R60702 and Gr70 plates were placed in a parallel direction. The spacers (R60702) were put on the surface of Gr70 to support the R60702 plate. The height of the spaces is about 8 mm. The ANFO explosive layer was put on the surface of the R60702 plate. The thickness of the explosive was 33-35 mm, the density was 0.75 g/cm³, and the velocity was 2300 m/s⁻¹. An electronic detonator was placed in the long edge of the plate.

By initiating the electronic detonator, the detonation propagated through the R60702 plate. This leads to partial melting of the wave crest material of the clad plate in the form of pulse. Once there is a defect such as the unbounded area in the pulse path, parts of the energy will be reflected back from the defect surface. In the present work, the UT records show that there is no area where one or more discontinuities produced a continuous total loss of back reflection pulse. Figure 3 is a UT record of the clad plate. There are the first back reflection and the second back reflection pulses in the clad plate. According to the ASTM A578 specification, it indicates the interface of the R60702-Gr70 clad plate is continuous. Thus, the bonding quality of the clad plate meets the requirement of ASTM A578.

After UT, microstructure analysis was implemented on the R60702 plate clad, Gr70 plate, and clad plate. Optical views of the microstructures of the clad plate are shown in Figures 4 and 5. Figure 4(a) shows the microstructure of the R60702 plate. It can be seen from the picture that zirconium alloy (R60702) consists of the α phase with a large grain size (about 20-40 μm). The size of α-Zr grains is about 8.5 level. The structure is made up of fully equal-axed grains. And there is no abnormal deposition in the metal structure. Figure 4(b) illustrates the microstructure of Gr70. This steel contains ferrite and pearlite grains with a band-like distribution.

### Table 1: The components of R60702.

| Material               | Hf   | Fe+C | C    | N    | H    | O    | Zr |
|-----------------------|------|------|------|------|------|------|----|
| ASTM SB551 R60702     | ≤4.5 | ≤0.1 | ≤0.05| ≤0.025| ≤0.005| ≤0.10| Bal.|
| Measured values       | 2.21 | 0.066| 0.007| 0.003| 0.001| 0.068| Bal.|

### Table 2: The components of Gr70.

| Material | C   | Si  | Mn  | P   | S   | Fe  |
|----------|-----|-----|-----|-----|-----|-----|
| SA 516 Gr70 | ≤0.20 | ≤0.035 | 1.20–1.60 | ≤0.015 | ≤0.015 | Bal. |
| Measured values | 0.18 | 0.015 | 1.14 | 0.005 | 0.008 | Bal. |
regions and to the formation of melt blocks (Figure 5(b), white arrow).

To further observe the microstructure characteristic of the welded interface, the SEM was employed to investigate the bonding interface as shown in Figure 6(a). And there are also no defects like pores and cracks on the surface of the clad plate. The results are in accordance with that of optical micrographic examination. And base on the results mentioned above, high-velocity collision leading to the temperature increase in local zones, melt blocks formed at the vortices in the clad surface (Figure 6(b)). The elemental components of the melt blocks were measured using EDX at the positions marked with white arrows 1, 2, and 3, in Figure 6(b). The results are shown in Figure 7. The spectrum positions in Figure 7(a) are corresponding to the arrows in Figure 6(b). As confirmed by the EDX analysis, the region at arrow 1 consists of 97.56% Fe, 2.06% Zr, and 0.38% Si; the region at arrow 2 consists of 100% Fe; and the region at arrow 3 consists of 44.51% Fe and 55.49% Zr (Figures 7(b) and 7(c)). Base on the EDX analysis results, it can be concluded that R60702 and Gr70 melted during the explosive welding, and the temperature at the welding interface exceeded the melting temperature of R60700 (1852°C) in a short time. The melt blocks resulted from the melting and mixing of the R60700 and Gr70 plates.

Because of the high temperature at the interface during the explosive welding, it could lead to the formation of the element diffusion across the interface [19–21]. However, Figure 6 indicates that there is no diffusion layer across the interface. In order to confirmation the element diffusion process (possible), line scanning of Fe and Zr, Cr, and Si elements are carried out away from 200μm across the interface. The results are shown in Figure 8. The R60700 and Gr70 materials contain different elements. At the bonding interface from the R60702 to Gr70, it can be seen from Figure 8(b) that the contents of Fe and Zr elements all show sharp transitions across the interface. It indicates that element diffusion during the explosive welding process is not observed under the accuracy of line scanning detection. And there is no silicon in R60702 material (Table 1), thus the silicon element in the specimen is from SiC sandpaper contamination during polishing.

Figures 9 and 10 show the mechanical properties of the R60702-Gr70 clad plate. The bending test was conducted to evaluate the bonding quality of the clad plate. Figure 9 shows the cross-sectional images of the R60702-Gr70 specimen.
after the bending tests. During the face-bend test, the R60702 plate was subjected to tensile stress, and the Gr70 plate was subjected to compressive stress. During the reversed-bend test, the R60702 plate was subjected to compressive stress, and the Gr70 plate was subjected to stress. Either face-bend specimen or reversed-bend specimen was not delaminated at the interface or parent metals. The result meets the requirements of the ASTM B898 specification.

The shearing test was also employed to evaluate the bonding quality of the clad plate. The shearing strength is 412 MPa. It is much higher than that of the ASTM 898 requirement (137 MPa). Figure 10 shows the shearing specimen after the shearing test. It can be seen that the separation appears at the interface. After the above examination, the clad plate was cut and rolled to the shell structure (shown in Figure 11). The diameter of the shell is φ508 mm. During the rolling, the clad plate was not delaminated at the interface or base metals. It indicates the clad plate meets the processing requirement.

4. Conclusion

An investigation on the R60702-Gr70 explosive clad plate was carried out to understand the microstructure and bonding properties. The important findings are as follows:

(1) The R60702-Cr70 clad plate with 3/25 × 2330 × 6000 mm was successfully obtained by the explosive welding technique. UT inspection indicates that the bonding quality meets the requirement of the ASTM A578 specification.
Figure 7: Continued.
The R60702-Gr70 explosive clad plate has a wavy interface. The wave is symmetrical. No cracks or pore defects were observed at the interface. And there is no element diffusion across the interface of the clad plate.

There are melt blocks on the interface. These melt blocks resulted from the melting and mixing of the R60702 and Gr70 materials during the high-velocity collision.

**TABLE 1**

| Spectrum | Si  | Fe  | Zr  | Total |
|----------|-----|-----|-----|-------|
| Spectrum 1 | 0.38 | 97.56 | 2.06 | 100.00 |
| Spectrum 2 | 0   | 100  | 0   | 100.00 |
| Spectrum 3 | 0   | 44.51 | 55.49 | 100.00 |

All results in weight%

**Figure 7:** SEM micrographs of the clad plate (a) and corresponding EDX spectrums: (b) white arrow 1 (spectrum1), (c) white arrow 2 (spectrum2), and (d) white arrow 3 (spectrum3).

**Figure 8:** The bonding interface in the R60702-Gr70 clad plate: (a) microstructure; (b) line scan analysis from R60702 to Gr70 plates.

**Figure 9:** The specimens after bending test: a) and b) outside bending (bended up to 105°), c) and d) inside bending (bended up to 180°).
The authors declare that (i) no support, financial or otherwise, has been received from any organization that may have an interest in the submitted work and (ii) there are no other relationships or activities that could appear to have influenced the submitted work.

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Data Availability

I declare that I do not have any commercial or associative interest that represents a conflict of interest in connection with the work submitted.

Conflicts of Interest

The authors declare that (i) no support, financial or otherwise, has been received from any organization that may have an interest in the submitted work and (ii) there are no other relationships or activities that could appear to have influenced the submitted work.

References

[1] J. L. Havenga and J. T. Nel, “The Manufacturing of Zirconium Metal Powder by Means of a High Temperature Plasma Process,” in 2010 Abstracts IEEE International Conference on Plasma Science, p. 1, Norfolk, VA, 2010.
[2] Y. Li and J. He, “Comparative and verified studies of zirconium nanocomposite nanofibres by bubble spinning,” Micro & Nano Letters, vol. 13, no. 2, pp. 228–231, 2018.
[3] I. V. Gayvoronsky and V. V. Girdzhon, “Structural state of zirconium surface layers after laser alloying by titanium and nickel,” 2015 International Young Scientists Forum on Applied Physics (YSF), Dnipropetrovsk, pp. 1–4, 2015.
[4] N. S. Pushilina, V. N. Kudiiarov, and A. N. Nikolaeva, “Investigation of hydrogen distribution in zirconium alloy by means glow discharge optical emission spectroscopy,” in 2014 9th International Forum on Strategic Technology (IFOST), pp. 489–491, Cox’s Bazar, 2014.
[5] S. Kaushik, K. Buddhadev, D. Sangita et al., “Direct non-destructive total reflection X-ray fluorescence elemental determinations in zirconium alloy samples,” Journal of Synchrotron Radiation, vol. 27, no. 5, pp. 1253–1261, 2020.
[6] L. L. Yang, M. H. Chen, J. L. Wang et al., “Microstructure and composition evolution of a single-crystal superalloy caused by elements interdiffusion with an overlay NiCrAlY coating on oxidation,” Journal of Materials Science & Technology, vol. 45, pp. 49–58, 2020.
[7] Y. X. Qiao, J. Huang, D. Huang et al., “Effects of laser scanning speed on microstructure, microhardness, and corrosion behavior of laser cladding Ni45 coatings,” Journal of Chemistry, vol. 2020, Article ID 1438473, 11 pages, 2020.
[8] A. Aushev, A. N. Balandina, E. N. Grishin, O. B. Drennov, A. M. Podurets, and M. I. Tkachenko, “Features of uniaxial and biaxial deformation of the contact boundary of metals during explosion welding,” Inorganic Materials: Applied Research, vol. 9, no. 3, pp. 504–508, 2018.
[9] M. Yang, H. M. Ao, S. Zhao, and J. F. Xu, “Application of colloid water covering on explosive welding of AA1060 foil to Q235 steel plate,” Propellants, Explosives and Pyrotechnics, vol. 45, no. 3, pp. 453–462, 2020.
[10] Z. R. Sun, C. Shi, F. Xu, F. Ke, and X. M. Wu, “Detonation process analysis and interface morphology distribution of double vertical explosive welding by SPH 2D/3D numerical simulation and experiment,” Materials & Design, vol. 191, pp. 108–117, 2020.
[11] E. V. Kuz’min, V. I. Lysak, S. V. Kuz’min, and M. P. Kerolev, “Effect of parameters of high-velocity collision on the structure and properties of joints upon explosive welding with simultaneous ultrasonication,” The Physics of Metals and Metallography, vol. 120, no. 2, pp. 197–203, 2019.
[12] T. T. Zhang, W. X. Wang, J. Zhou et al., “Investigation of interface bonding mechanism of an explosively welded tri-metal titanium/aluminum/magnesium plate by nonindentation,” JOM: the journal of the Minerals, Metals & Materials Society, vol. 70, no. 4, pp. 504–509, 2018.
[13] S. Y. Yang and J. W. Hao, “Microstructure and properties of 5083 Al/1060 Al/AZ31 composite plate fabricated by explosive welding,” Journal of Materials Engineering & Performance, vol. 27, no. 3, pp. 1177–1184, 2018.
[14] J. Loiseau, W. Georges, D. L. Frost, and A. J. Higgins, “The propulsive capability of explosives heavily loaded with inert materials,” Shock Waves, vol. 28, no. 4, pp. 709–741, 2018.
[15] P. W. Chen, J. R. Feng, Q. Zhou et al., “Investigation on the explosive welding of 1100 aluminum alloy and AZ31 magnesium alloy,” *Journal of Materials Engineering and Performance*, vol. 25, no. 7, pp. 2635–2641, 2016.

[16] *Standard specification for straight-beam ultrasonic examination of rolled steel plates for special applications*, pp. 1–5, 2017, ASTM A578.

[17] *Standard specification for reactive and refractory metal clad plate*, pp. 1–9, 2016, ASTM B898.

[18] R. C. Gupta and G. S. Kainth, “Swinging wake mechanism for interface wave generation in explosive welding of metals,” *Journal of Applied Mechanics*, vol. 57, no. 3, pp. 514–521, 1990.

[19] S. P. Kiselev and V. I. Mali, “Numerical and experimental modeling of jet formation during a high-velocity oblique impact of metal plates,” *Combustion Explosion & Shock Waves*, vol. 48, no. 2, pp. 214–225, 2012.

[20] S. Saravanan and K. Raghukandan, “Thermal kinetics in explosive cladding of dissimilar metals,” *Science and Technology of Welding and Joining*, vol. 28, no. 5, pp. 99–103, 2013.

[21] T. Onzawa and Y. Ishii, *Fundamental studies on explosive welding. welding research abroad*, vol. 23, no. 3, pp. 28–36.