Magnetic order of tetragonal CuO ultrathin films

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We present a detailed low-energy muon spin rotation and x-ray magnetic circular dichroism (XMCD) investigation of the magnetic structure in ultrathin tetragonal (T)-CuO films. The measured muon-spin polarization decay indicates an antiferromagnetic (AFM) order with a transition temperature higher than 200 K. The XMCD signal obtained around the Cu $L_{2,3}$ edges indicates the presence of pinned Cu$^{2+}$ moments that are parallel to the sample surface, and additionally, isotropic paramagnetic moments. The pinning of some of the Cu moments is caused by an AFM ordering, which explains these experimental findings, showing that the magnetic order of the T-CuO ultrathin films differs from the theoretical predictions.

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I. INTRODUCTION

The discovery of high-temperature superconducting cuprates (HTSCs) [1,2] has increased the attention on materials based on different Cu-O building blocks. It was realized that electron correlations in two-dimensional materials based on different Cu-O building blocks. It was realized that electron correlations in two-dimensional materials based on different Cu-O building blocks. A higher-symmetric tetragonal distorted rocksalt form of CuO (T-CuO) has been synthesized as ultrathin films of approximately six unit cells (u.c.) grown on top of SrTiO$_3$ substrate [5]. In this material, CuO$_6$ octahedra form two-dimensional (2D) CuO planes with both edge- and corner-sharing Cu-O bonds that are stacked along the crystal c-axis (Fig. 1). At variance with the CuO$_2$ cuprate layers, oxygen ions do not bridge nearest-neighbor (NN) but next-NN copper ions [6,7]. The electronic structure of T-CuO is determined by the properties of its 2D edge-sharing lattice planes, each conceptually built from two interpenetrating corner-sharing sublattices, with lattice parameters typical of HTSCs and $180^\circ$ Cu-O-Cu bonds. A small coupling between these frustrated sublattices gives rise to slight quasiparticle renormalization [8]. In addition, T-CuO extends the simple rocksalt transition-metal oxide series from FeO, CoO, NiO to CuO, and it has been predicted to have a very high magnetic ordering temperature, with Néel temperature ($T_N$) close to 800 K [9,10]. Therefore, T-CuO is a very useful model system to test theoretical first-principles calculations on magnetic interactions. A study using a CuO-Cu$_2$O heterostructure with a highly symmetric CuO phase claims to find an antiferromagnetic order below $T_N = 600$ K [11]. Nevertheless, the study in Ref. [11] has a clear shortcoming in that it claimed to be unable to distinguish between the two structural phases Cu$_2$O and cubic CuO through an XRD (x-ray powder diffraction) analysis. The lattice parameter for Cu$_2$O is approximately 4.28 Å [12] and the lattice parameter for the cubic CuO phase can be estimated from the average lattice parameters of the T-CuO synthesized by Siemons [5], which is approximately 4.38 Å. The 0.1 Å lattice parameter difference would be easily detected and sufficient to distinguish between the two structures in an XRD analysis for angles larger than 50 °. However, the work published in the literature [11] does not find such a contribution, and therefore the presence of a cubic CuO phase has not been demonstrated, which makes the discovery related to $T_N$ ambiguous.

Experimentally, resonant inelastic x-ray scattering (RIXS) measurements reported spin-wave excitations dispersing through two cupratelike antiferromagnetic (AFM) sublattices. Two possible scenarios for the magnetic structure have been proposed with AFM orderings along the film plane [8].

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However, as of yet there is no direct experimental confirmation of a magnetic order in T-CuO and its possible spin structure, since magnetic excitations away from the zone center have also been found in superconducting cuprates by RIXS at doping levels far from the antiferromagnetically ordered state [14]. Here, we perform a series of complementary low-energy muon spin rotation (LE-μSR) and x-ray magnetic circular dichroism (XMCD) measurements on T-CuO ultrathin films. Our results present clear evidence of static magnetic order in T-CuO. XMCD finds pinned magnetic moments, which point to an underlying AFM as well as some paramagnetic contribution from the interface or surface region. These pinned moments clearly suggest an AFM ordering that is preserved above room temperature, but they are difficult to understand in the context of the predicted magnetic ordering schemes of T-CuO.

II. EXPERIMENTS

A. Sample preparation

Four nominally identical T-CuO thin films have been epitaxially grown on the (001) surface of a SrTiO$_3$ (STO) substrate with dimensions 10 × 10 mm by pulsed laser deposition, following the same procedure reported in Ref. [5]. The T-CuO layer thickness was estimated to be 3.6 nm (approximately 6 u.c.), in which it was controlled by reflection high-energy electron diffraction. A thin carbon layer of 10 nm thickness was deposited on top of the T-CuO films at room temperature and at 10$^{-5}$ mbar pressure by a commercial LEICA EM ACE600 high vacuum sputter coater, which measures in situ the thickness of the capping C layer. These samples are referred to as C/T-CuO/STO. The carbon layer serves as a protection layer for the T-CuO without affecting the properties of the thin film. Additionally, four (001)-STO bare substrates of 10 × 10 mm were coated in the same way and used as reference. The coated substrates are labeled as C/STO.

B. Low-energy muon spin rotation

Low-energy muon spin rotation (LE-μSR) experiments have been performed in the LEM spectrometer [13–15] at the Paul Scherrer Institute. One of the great advantages of the LEM spectrometer is that it can implant 100% spin-polarized positively charged muons (μ$^+$) in the investigated material. Due to the very thin layer of T-CuO, implantation energies from 1 to 6 keV were selected for this investigation. A static magnetic field of 100 G, perpendicular to the sample surface, and initial muon polarization were applied during the measurements. An array of eight positron detectors placed around the sample measures the decay product of the muons, i.e., positrons, as a function of time. The asymmetry in the number of detected positrons between two opposite detectors is proportional to the spin polarization of the muons along their axis [16]. For the LE-μSR experiments, the four C/T-CuO/STO samples were arranged in a mosaic covering an area of 20 × 20 mm and glued using silver paint onto a silver-coated aluminum plate. The same procedure was carried out with the four C/STO reference samples. The plate was then mounted onto a cold finger of a He flow cryostat, which operates in the temperature range from 5 to 300 K. Due to the large beam area at the LEM spectrometer, only $\sim$75% of the muons stop in the sample itself while $\sim$25% stop in the silver coating and contribute a nonrelaxing background signal.

C. X-ray magnetic circular dichroism

The x-ray magnetic circular dichroism (XMCD) data were collected at the X-TREME beamline [17] of the Swiss Light Source of the Paul Scherrer Institute. An x-ray beam with a spot size of 1 × 1.3 mm was used. The setup at X-TREME allows us to field-cool (FC) the sample from 360 to 3 K in a magnetic field ranging from 6.5 T (+FC) to −6.5 T (−FC). The magnetic field was always parallel to the incident x-ray beam, and the direction of the applied magnetic field preceded the FC direction. Incident angles for the x-rays of 30° and 90° from the sample surface were selected, ensuring...
enhancements of the in-plane and out-of-plane components of the magnetization in the XMCD signal, respectively. Measurements collected in total electron yield mode were carried out with the photon energies tuned around the Cu L2,3 absorption edge. One of the C/T-CuO/STO samples used for the LE-μSR experiments was used for the XMCD measurements. Additionally, x-ray linear dichroism (XLD) measurements were also performed at room temperature and with a 30° incident angle.

III. RESULTS

A. Low-energy muon spin rotation

Low-energy muons are highly sensitive probes of the local magnetic fields in the investigated sample. Thanks to their tunable implantation energy (E), and corresponding implantation depth on nanometer scale, the muons can be used to study very thin samples. The implanted muons decay emitting positrons (e+), preferentially, in the muon spin direction at the time of decay. By detecting the time dependence of the positron counts in different detectors \( N_{e^+}(t) \), we can determine the spin polarization of the muon \( |P(t)| \) as [18]

\[
N_{e^+}(t) = N_0(1 + A_\mu P(t)e^{-t/t_{\tau_\mu}}),
\]

where \( N_0 \) is the total number of recorded positrons at \( t = 0 \), and \( A_\mu \) is the asymmetry parameter, which depends on the anisotropy of the beta decay of the \( \mu^+ \) and the geometry of the detector. The exponential decrease in \( N_{e^+}(t) \) is due to the muon lifetime \( t_{\tau_\mu} = 2.2 \mu s \). In the presence of a constant magnetic field, applied transverse to the initial muon spin polarization, \( P(t) \) precesses due to the influence of the magnetic field following

\[
A_\mu P(t) = A_\mu e^{-t^2/t_{\tau_\mu}^2} \cos(\omega_{L}\mu t + \phi).
\]

The Larmor frequency is \( \omega_{L} = \gamma_\mu B \), where \( \gamma_\mu = 0.8516 \) Mrad/mT is the muon gyromagnetic ratio and \( B \) is the local magnetic field detected by muons, which corresponds to the sum of the local magnetic field of the sample and the external applied magnetic field. The Gaussian envelope in Eq. (2) represents the damping of the oscillating polarization with relaxation parameter \( \sigma \) due to the width of the local field distribution. The phase \( \phi \) depends on the relative orientation of the initial polarization and the positron detector. In insulating materials, muonium formation occurs when a muon binds an electron forming a hydrogenlike state \( (\mu^+ + e^-) \).

To determine and optimize the fraction of muons implanted in the very thin T-CuO film, TRIM.SP Monte Carlo simulations [18,19] were performed (Fig. 2). Figure 2 shows calculations with an implantation energy of 1.5 keV, a 10-nm carbon-capping layer on top of T-CuO deaccelerates the muons and results in an optimized stopping fraction of muons in T-CuO of 20–30%. For this purpose, the four T-CuO thin films have been coated with a 10 nm carbon layer (C/T-CuO/STO) and additionally the reference C/STO samples.

The asymmetry as a function of temperature was measured in both samples at various implantation energies. The data have been fitted using Eq. (2) to evaluate the relevant parameters. Figure 3(a) displays the temperature dependence of the \( A_\mu \) measured in the C/T-CuO/STO sample for different implantation energies. At 5 keV, the temperature dependence clearly resembles that observed in bare STO [20]. The variation in temperature around 90 K is attributed to the muonium formation in the STO substrate that accounts for the “lost” diamagnetic asymmetry at low temperatures. With decreasing \( E \), we expect an increase in the contribution from the T-CuO layer. Data with an implantation energy of 2 keV show that the signal from the bare STO still remains. At 1.5 keV a clear deviation from bare STO becomes more apparent, as expected from the TRIM.SP simulation (Fig. 2). This difference is clearly seen when comparing \( A_\mu \) measured in C/STO and C/T-CuO/STO as a function of implantation energy, as shown in Fig. 3(b). Note that these measurements were performed at high temperatures, where there is no muonium formation in STO. At high implantation energy (i.e., when muons stop in the substrate), the asymmetry measured in both samples is equal, as expected for samples of similar size and composition. However, with decreasing \( E \), a gradual difference between the asymmetries of both samples is observed, where C/T-CuO/STO exhibits a larger reduction. We attribute the reduced asymmetry in C/T-CuO/STO to a fast depolarization of muons implanted in the T-CuO layer. This fast depolarization

![FIG. 2. Results of TRIM.SP simulation to calculate the stopping distribution of muons in C/T-CuO/STO. (a) Stopping profile as a function of depth for different implantation energies. (b) Stopped fraction of muons, for each layer, as a function of implantation energy.](image-url)
is a strong indication of a large local magnetic field present in this layer, which is already present at 200 K.

Figures 3(c)–3(e) display the parameters $A_{\mu}$, the mean field $B$, and the damping rate $\sigma$ extracted from the fits using Eq. (2) as a function of temperature for the C/STO and the C/T-CuO/STO samples at $E=1.5$ keV. The asymmetry in Fig. 3(c) for the C/T-CuO/STO sample reduces by approximately 6% compared to the C/STO in the full temperature range. This reduction is consistent with the energy dependence displayed in Fig. 3(b) and presents a clear sign of magnetic order in T-CuO layer. The mean field $B$ [Fig. 3(d)] exhibits no temperature dependence and no significant difference between the C/T-CuO/STO and the C/STO samples. In contrast, the damping rate $\sigma$ [Fig. 3(e)] shows a small enhancement in C/T-CuO/STO at low temperatures compared to C/STO.

To understand the nature of magnetic order in the T-CuO layer, we note first that the fraction of muons stopping in the T-CuO predicted by the TRIM.SP simulations is $\sim 25\%$ at 1.5 keV, as shown in Fig. 2. Considering that only $\sim 75\%$ of the muons contribute to the signal from the C/T-CuO/STO sample (the remaining $\sim 25\%$ stop in the silver coating) and assuming that all the muons implanted in the T-CuO experience a large magnetic field and depolarize, we would expect a loss of asymmetry of $\sim 18\%$ in C/T-CuO/STO compared to C/STO. However, we observe approximately 6% symmetry reduction, which is just 1/3 of the asymmetry reduction expected. This indicates that only 1/3 of the T-CuO volume exhibits long-range magnetic order. Nevertheless, this estimation is very crude since it is based on TRIM.SP calculations and takes the bulk densities of C and T-CuO, which might be significantly different in the studied heterostructure.

B. X-ray magnetic circular dichroism

XMCD is the difference between the absorption spectra taken with right and left circularly polarized light (C+ and C−). The dichroic signal is nonzero for systems with a net magnetization, such as ferromagnets, uncompensated AFM, paramagnets in applied fields, etc. In addition, XMCD is directly proportional to the magnetic moment component that is parallel to the x-ray beam direction. It should be noticed that a perfect compensated AFM system will therefore not create an XMCD signal, since the magnetic moment is fully compensated. However, we found a clear XMCD signal, as our following results show.

Figure 4 shows the x-ray absorption spectrum (XAS) and the XMCD signal for +FC and an applied field of 6.5 T in
FIG. 4. XAS and XMCD signals at 3 K for positive FC (6.5 T) and applied magnetic field of (a),(b) 6.5 T and (c),(d) −6.5 T, respectively. Data have been collected at 30 ° incident angle. The dashed green lines indicate the background subtracted from the XAS signals. The integrated XAS is calculated as \( \int \text{XAS} \). panels (a) and (b) and −6.5 T in panels (c) and (d), respectively, both taken at 3 K and covering the energy range of the Cu \( L_{2,3} \) edges. The XAS has been represented as \( (\mu^+ + \mu^-)/2 \) and the electron transitions to the continuum by a two-step function. The XMCD signal is \( (\mu^+ - \mu^-) \). The measurements were performed with an incident angle of 30 °. We detect a much larger XMCD signal for the (b) case with 6.5 T applied compared to the (d) case with −6.5 T applied. This indicates that a fraction of magnetic Cu\(^{2+} \) moments pointing parallel to the \(+H\) direction do not reverse their orientation in the opposite magnetic field at this temperature. This can be attributed to pinned moments [21]. In addition, we observe a relative increase of the \( L_2/L_3 \) XMCD intensity ratio for the (d) case. Note that the XAS signals [panels (a) and (c)] around 932 eV show a small production of Cu\(^{1+} \) ions caused by the x-ray beam damage. The production of Cu\(^{1+} \) ions has been minimized by enlarging the spot size of the x-ray beam and therefore reducing the x-ray beam density and, when possible, changing the beam position on the sample surface every 20 min.

One can estimate, through XMCD sum rules, the orbital \( (m_{\text{orb}}) \) and spin moment \( (m_{\text{spin}}) \) per Cu ion of our system as [22,23]

\[
\begin{align*}
    m_{\text{spin}} &= -\frac{3p - 2q}{r} \left(1 + \frac{7T_z}{2 \langle S_z \rangle} \right)^{-1}, \\
    m_{\text{orb}} &= -\frac{2q}{3r} N_h, 
\end{align*}
\]

where \( p \) is the integral of the XMCD signal over the Cu \( L_3 \) edge, \( q \) is the integral of the XMCD signal over the \( L_3 \) and \( L_2 \) edges, and \( r \) is the integral of the isotropic spectra over \( L_{2,3} \) edges subtracting the step function; all parameters are indicated in Fig. 4. \( T_z \) is the dipolar term and \( N_h = 1 \) is the number of holes in the 3\( d \) shell for the Cu\(^{2+} \) state. In cases in which the ion is in a high-symmetry site like an octahedral site, \( T_z \)
The integrated XAS is calculated as
\[ \int \text{respectively. Data have been collected at 30° incident angle. The dashed green lines indicate the background subtracted from the XAS signals.} \]

\[ \text{a 10% uncertainty as recommended in Ref. [35].} \]

For T-CuO extracted from XMCD sum rules around the Cu atoms, it might not be negligible [24]. We calculated the crystal-field parameters using the CRISPY software [25] that is based on the QUANTY code [26]. The x-ray linear dichroism (XLD) spectra are in good agreement with our experimental XLD data (Fig. 5). The XLD is the difference between horizontal and vertical linear polarization, and it is very sensitive to the crystal-field splitting. From the calculation we obtain a value of \( Tz \) = 0.016, \( Sz \) = -0.494, and \( Lz \) = -0.121. The total magnetic moment \( (Sz + Lz) \) obtained by multiplet calculations is of comparable size to those reported by cupric oxides systems [4]. The XMCD sum rule results using Eqs. (3) and (4) are listed in Table I.

We repeat the XMCD measurements using opposite field cooling in -6.5 T from 360 to 3 K, which results in the spectra of Fig. 6. The XMCD signal maintains the same orientation as those found in the previous case of +FC (Fig. 4). However, the XMCD signal for Fig. 6(b) with 6.5 T applied field remains larger than the (d) case with -6.5 T applied field, but the difference between (b) and (d) is smaller in size compared to the +FC case [Figs. 4(b)–4(d)]. This indicates that some pinned moments still point along the +H direction, i.e., they cannot be reversed even at high temperatures. The corresponding sum-rule results for the –FC case are also shown in Table I.

The absolute value for the spin moment obtained through XMCD sum rules is approximately 0.1 \( \mu_B \)/atom for both opposite FC cases. However, the absolute value of the orbital magnetic moment for the +FC case differs by approximately a factor of 1.5 between 6.5 and -6.5 T. This is similar for the –FC case. Nevertheless, we find that the total magnetic moments are close to each other in all cases with approximately 0.12 \( \mu_B \)/atom. Moreover, we note that the spin and orbital moments conserve the same sign for the same field applied independently of the FC direction.

Figure 7 shows the XMCD signal taken at the maximum of the Cu \( L_3 \) edge as a function of applied magnetic field following the two opposite FC processes. The XMCD signal is independent of the field ramping direction and does not show any hysteresis, indicative of an absence of an intrinsic ferromagnetic state. The XMCD signal exhibits a clear

| TABLE I. Orbital, spin, and total magnetic moment of Cu\(^{2+}\) in T-CuO extracted from XMCD sum rules around the Cu \( L_{2,3} \) edges at 3 K. FC from 360 to 3 K in a field of either 6.5 or -6.5 T and data taken in applied field of 6.5 or -6.5 T. The XMCD sum rules include a 10% uncertainty as recommended in Ref. [35]. |
|---------------------------------|---------------------------------|---------------------------------|
| \( H = 6.5 \text{T(} \mu_B/\text{atom)} \) | \( H = -6.5 \text{T(} \mu_B/\text{atom)} \) |
| \(+\text{FC (6.5 T)} \)          | \(-\text{FC (-6.5 T)} \)          |
| \( m_{\text{orb}} \)             | \( m_{\text{orb}} \)             |
| \( m_{\text{spin}} \)            | \( m_{\text{spin}} \)            |
| \( M_{\text{total}} \)           | \( M_{\text{total}} \)           |
| \( m_{\text{orb}} = -0.035 \pm 0.003 \) | \( m_{\text{orb}} = 0.024 \pm 0.002 \) |
| \( m_{\text{spin}} = -0.11 \pm 0.01 \) | \( m_{\text{spin}} = 0.11 \pm 0.01 \) |
| \( M_{\text{total}} = -0.14 \pm 0.01 \) | \( M_{\text{total}} = 0.13 \pm 0.01 \) |
| \( m_{\text{orb}} = -0.015 \pm 0.001 \) | \( m_{\text{orb}} = 0.009 \pm 0.001 \) |
| \( m_{\text{spin}} = -0.10 \pm 0.01 \) | \( m_{\text{spin}} = 0.11 \pm 0.01 \) |
| \( M_{\text{total}} = -0.11 \pm 0.01 \) | \( M_{\text{total}} = 0.12 \pm 0.01 \) |

FIG. 6. XAS and XMCD signals at 3 K for negative FC (-6.5 T) and applied magnetic field of 6.5 T in (a),(b) and -6.5 T in (c),(d), respectively. Data have been collected at 30° incident angle. The dashed green lines indicate the background subtracted from the XAS signals. The integrated XAS is calculated as \( / \text{XAS-step function} \).
positive shift in $y$-axis from the origin. The shift indicates that a significant fraction of the magnetic moments do not reverse their orientation with the applied magnetic field at low temperature, a clear indication that these moments remain pinned. The estimated amount of pinned moments is approximately 37\% of the total moment ($M_{\text{pin}} + M_{\text{rot}}$) observed in the $+\text{FC}$ case [Fig. 7(a)]. The measured data are compared with the calculated magnetization (dashed blue curve) based on a paramagnetic system with spin $\frac{1}{2}$ in an applied field at 3 K without any pinned moment contribution. The calculation represents simply the Brillouin function [27]. The excellent agreement of our data removing the pinned contribution with this simple model indicates a mixture of paramagnetic and pinned magnetic Cu moments.

Figure 7(b) shows the results for the $-\text{FC}$ case. Interestingly, again a positive shift in $y$-axis is observed, although significantly smaller, reflecting the smaller contribution from pinned moments that account for 16\% of the total observed moment. This indicates that not all moments that are pinned at low temperature reverse at 360 K in $-6.5$ T field, and those pinned moments that do not reverse have a preferred orientation.

In addition, Fig. 8 shows the temperature dependence of the XMCD signal at the maximum of the Cu $L_3$ edge. We find a reasonable agreement between the partition function $Z$ of a paramagnetic system with spin $\frac{1}{2}$ [27].

To study the direction of the pinned moments, Fig. 9 displays an XMCD hysteresis loop performed at normal incidence angle. The sample was FC in $-6.5$ T from 360 to 3 K. Contrary to the previous case, Fig. 9 does not exhibit a vertical shift, indicating that the pinned moments do not have any out-of-plane contribution, and therefore the XMCD signal consists of only the paramagnetic contribution. This confirms that the pinned moments are parallel to the sample surface. Moreover, the paramagnetic signal is of comparable size to that probed at $30^\circ$ grazing angle [Fig. 7(a)], which shows that the paramagnetic spins are isotropic. However, the field dependence shows a flattening for large positive fields that might be somehow related to the occurrence of pinned moments.

Furthermore, to identify the contribution of pinned moments to the signal, XMCD measurements in zero-field are shown in Fig. 10. These measurements indicate that the pinned moments have a larger orbital contribution compared to the spin contribution; a similar case was reported in [28,29]. Applying the sum rules, we find that the pinned moments exhibit a ratio of $m_{\text{orb}} / m_{\text{spin}} = 1.5$ and a total magnetic moment of $0.017 \pm 0.001 \mu_b/\text{atom}$. Comparing these values to the total magnetic signal observed at applied field measurements, $M_{\text{total}} = 0.12 \mu_b/\text{atom}$ (Table I), we deduce that the pinned...
moments represent approximately 15%, in agreement with our above estimation from Fig. 7(b).

As the paramagnetic and the pinned moments are most likely located at the interfaces, it would be interesting to check if there is a magnetic moment at the Ti ions from the STO substrate. Such moments have been already reported in the presence of a two-dimensional electron gas (2DEG) [30–33]. The XAS and the XMCD signals obtained at the Ti $L_{2,3}$ edges are shown in Fig. 11. Only a very small XMCD signal is observed around the Ti $L_{2,3}$ edges, which is similar to those found in annealed LAO/STO samples [30]. The detected XMCD signal is believed to originate from the Ti$^{4+}$($3d^0$) ions related to the magnetic environment sensitivity of the electrons excited to the $2p^5d^1$ state, as has been previously reported, supported by multiplet calculations and discussed in Refs. [30,33]. However, we do not observe any response from magnetic Ti$^{3+}$ ions similar to that found for the nonannealed LAO/STO [31] or in $\gamma$-Al$_2$O$_3$/SrTiO$_3$ [33], both exhibiting a 2DEG. This indicates that the T-CuO/STO interface is unlikely to host a 2DEG gas and that the observed XMCD is not due to an accumulation of Ti$^{3+}$ or any associated magnetic order at the STO interface as reported in other systems [30–33].

### IV. DISCUSSION AND CONCLUSIONS

The relatively large fraction of depolarized muons implanted in the C/T-CuO/STO sample compared to the C/STO sample throughout the investigated temperature range gives a clear indication that a significant portion of the C/T-CuO/STO film is strongly magnetic. The 6% reduction of asymmetry in comparison with C/STO corresponds to at least $1/3$ of the volume fraction contribution of the muons stopping in the T-CuO layer, predicted from our very rough TRIM.SP calculations ($\sim 18\%$). This indicates that at least $1/3$ of the T-CuO volume has a long-range magnetic order. This is also supported by our depth-dependent measurements in Fig. 3(b). It is also important to point out here that the lost asymmetry does not exceed what we expect from the T-CuO layer, evidencing that the magnetic order in this layer cannot be ferromagnetic. Strong static stray fields from a thin ferromagnetic layer would have caused a much larger loss of asymmetry, larger than the volume fraction of the layer itself [34]. In addition, the absence of a significant magnetic moment at the Ti ions on the STO substrate also supports the idea of a strongly magnetic T-CuO layer.
FIG. 12. Sketch of the proposed model for T-CuO. Part (a) represents the + FC case, and (b) represents the −FC case. The blue layer highlights the paramagnetic moments, the green layer corresponds to the pinned moments, and the orange layer corresponds to the antiferromagnetic moments. Note that in (a) the pinned moment layer is enhanced because the pinned moment and paramagnetic moments (PM) have the same orientation.

of a noticeable magnetic field shift in Fig. 3(d) confirms that T-CuO cannot be ferromagnetic, since this would lead to a negative shift relative to the nonmagnetic C/STO [36]. Finally, the enhancement of $\sigma$ is another indication of the magnetic field fluctuations experienced by muons stopping in the vicinity of the T-CuO layer. These observations together suggest a scenario in which the T-CuO layer orders antiferromagnetically (or a magnetic order with very small net magnetic moment), and they constitute a direct observation of the magnetic order in T-CuO.

The field-dependent XMCD data of T-CuO exhibit two magnetic contributions, one coming from isotropic paramagnetic Cu$^{2+}$ moments and another from pinned moments lying in the sample surface plane. We have observed that the pinned moment contribution depends strongly on the field direction in the FC procedure starting at 360 K. This indicates that the pinned moment contribution originates in two portions/fractions: “permanently” and “eventually” pinned moments. The “permanently” pinned moments remain even at 360 K and the “eventually” pinned moments are reversible at 360 K but are frozen at 3 K. The XMCD sum rules give an estimation of the total magnetic moment of around 0.12 $\mu_B$/atom, from which approximately 15% of the total magnetic moment corresponds to permanently pinned moments. The contribution of pinned moments can be enhanced up to 37% by the “eventually” pinned moments depending on the FC direction. The observed Cu$^{2+}$ magnetic moment of 0.12 $\mu_B$/atom is only 1/5 of the total magnetic moment obtained by the multiplet calculations ($S_z + L_z = 0.615 \mu_B$/atom). Taking into account that the T-CuO layer consists of only 6 u.c., the magnetic signal observed comes from just approximately 1.2 u.c. of T-CuO.

The simplest model to explain our observations is schematically presented in Fig. 12. It is common that an AFM system shows uncompensated moments at the surface, e.g., due to uncompensated u.c., surface roughness, and/or defects and domain walls. All of them lead to missing magnetic exchange paths from the discontinuity of the magnetic ordering. We have shown that T-CuO exhibits uncompensated moments and with its major fraction being paramagnetic (PM) even at low temperature. Our XMCD data indicate that the pinned moments have an enhanced orbital moment contribution, which one could expect from inversion symmetry breaking or reduced dimensionality, e.g., as found for increased angular magnetic moments at a surface [34]. In addition, we have demonstrated that the pinned moments are oriented along the sample surface. One would assume that the orientation of the pinned moments reveals the direction of the AFM ordering. That would imply an AFM ordering of ferromagnetic planes along (001). The FC process enhances the pinned moment layer when it is along the pinned moment direction. The fact that a significant fraction of the pinned moments has the same preferred orientation (i.e., “permanently” pinned moments), independently of the direction of FC applied after warming up to 360 K, is a clear indication that the AFM ordering is present even at 360 K, which represents a lower limit for the transition temperature in T-CuO.

Although our simplistic model explains all the phenomena observed in our study, an AFM ordering of FM planes along the (001) direction does not agree with the theoretical prediction in Ref. [8]. The magnetic exchange interaction between the Cu$^{2+}$ ions avoids having ferromagnetic single layers as the orange layer sketched in our model (Fig. 12). Nevertheless, T-CuO is highly frustrated, making it more difficult to predict theoretically the nature of magnetic ordering. In addition, the missing magnetic exchange interactions from defects at the surface call into question the assumption that the pinned moments are parallel to the AFM moments. However, the fraction of pinned moments (16% of the total magnetic moment observed) is significant, and it might be difficult to solely attribute them to defects.

Summarizing, our LE-$\mu$SR study supports an AFM ordering in T-CuO with a transition temperature higher than 200 K. The XMCD investigation exhibits a magnetic signal constituted by pinned moments oriented along the sample surface and isotropic paramagnetic moments. The origin of the isotropic paramagnetic moments is most likely roughness, surface defects, and/or uncompensated spins. In addition, the
detection of pinned moments lying in the film plane and remaining up to 360 K gives a clear indication of a bias that originates from an underlying AFM order.

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[1] J. G. Bednorz and K. A. Müller, Possible high Tc superconductivity in the Ba-La-Cu-O system, Z. Phys. B 64, 189 (1986).
[2] M. K. Wu, J. R. Ashburn, and C. J. Torng, Superconductivity at 93 K in a New Mixed-Phase Y-Ba-Cu-O Compound System at Ambient Pressure, Phys. Rev. Lett. 58, 908 (1987).
[3] H. Eskes, L. H. Tjeng, and G. Sawatzky, Cluster-model calculation of the electronic structure of CuO: A model material for the high-Tc superconductors, Phys. Rev. B 41, 288 (1990).
[4] B. X. Yang, T. R. Thruston, J. M. Tranquada, and G. Shirane, Magnetic neutron scattering study of single-crystal cupric oxide, Phys. Rev. B 39, 4343 (1989).
[5] W. Siemons, G. Koster, D. H. A. Blank, R. H. Hammond, T. H. Geballe, and M. R. Beasley, Tetragonal CuO: End member of the 3d transition metal monoxide, Phys. Rev. B 79, 195122 (2009).
[6] S. Moser et al., Angle-Resolved Photoemission Spectroscopy Of Tetragonal CuO: Evidence for Intralayer Coupling Between Cuprate Like Sublattices, Phys. Rev. Lett. 113, 187001 (2014).
[7] I. J. Hamad, L. O. Manuel, and A. Aligia, Generalized One-Band Model Based on Zhang-Rice Singlets for Tetragonal CuO, Phys. Rev. Lett. 120, 177001 (2018).
[8] S. Moser, N. E. Shaik, D. Samal, S. Fatale, M. Dantz, J. Pelliciari, P. Olalde-Velasco, T. Schmitt, G. Koster, F. Mila, H. M. Ronnow, and M. Grioni, Magnons in tetragonal CuO, Phys. Rev. B 92, 140404(R) (2015).
[9] G. Peralta, D. Puggioni, A. Filippetti, and V. Fiorentini, Jahn-Teller stabilization of magnetic and orbital ordering in rocksalt CuO, Phys. Rev. B 80, 140408(R) (2009).
[10] X.-Q. Chen, C. L. Fu, C. Franchini, and R. Podloucky, Hybrid density-functional calculation of the electronic and magnetic structures of tetragonal CuO, Phys. Rev. B 80, 094527 (2009).
[11] K. S. Rabinovich, L. L. Samoilenko, A. S. Zhuravleva, and A. G. Shneider, Magnetic properties of high-symmetry CuO, App. Phys. Lett. 104, 182406 (2014).
[12] DOE Data Explorer, Materials Data on Cu2O by Materials Project (Web, USA, 2020), doi: 10.17188/1207131.
[13] T. Prokscha, E. Morenzoni, K. Deiters, F. Foroughi, D. George, R. Kobler, A. Suter, and V. Vranckovic, The new μE4 beam at PSI: A hybrid-type large acceptance channel for the generation of a high intensity surface-muon beam, NIMPRS A: Accelerators, Spectrometers, Detectors Assoc. Equip. 595, 317 (2018).
[14] M. Le Tacon et al., Intense paramagnon excitations in a large family of high-temperature superconductors, Nat. Phys. 7, 725 (2011).
[15] E. Morenzoni et al., Low-energy μSR at PSI: Present and future, Physica B 289-290, 653 (2000).
[16] A. Yau and P. Dalmas de Reutier, Muon Spin Rotation, Relaxation and Resonance: Applications to Condensed Matter (Oxford University Press, Oxford, 2011).
[17] C. Piamonteze et al., X-Treme beamline at SLS: X-ray magnetic circular and linear dichroism at high field and low temperature, J. Synchrotron Rad. 19, 661 (2012).
[18] E. Monzeneri, H. Glueckler, T. Prokscha, R. Khasanov, H. Luetkens, M. Birke, E. M. Forgan, Ch. Niedermayer, and M. Pleines, Implantation studies of keV positive muons in thin metallic layers, NIM B 192, 254 (2002).
[19] W. Eckstein, Computer Simulation of Ion–Solid Interactions (Springer, Berlin, 1991).
[20] Z. Salman, T. Prokscha, A. Amato, E. Morenzoni, R. Scheuermann, K. Sedlak, and A. Suter, Direct Spectroscopic Observation of a Shallow Hydrogenlike Donor State in Insulating SrTiO3, Phys. Rev. Lett. 113, 156801 (2014).
[21] H. Ohldag, A. Scholl, F. Nolting, E. Arenholz, S. Maat, A. T. Young, M. Carey, and J. Stohr, Correlation Between Exchange Bias and Pinned Interfacial Spins, Phys. Rev. Lett. 91, 017203 (2003).
[22] B. T. Thole, P. Carra, F. Sette, and G. van der Laan, X-Ray Circular Dichroism as a Probe of Orbital Magnetization, Phys. Rev. Lett. 68, 1943 (1992).
[23] C. T. Chen, Y. U. Idzerda, H. J. Lin, N. V. Smith, G. Meigs, E. Chaban, G. H. Ho, E. Pellegrin, and F. Sette, Experimental Confirmation of the x-Ray Magnetic Circular Dichroism Sum Rules for Iron and Cobalt, Phys. Rev. Lett. 75, 152 (1995).
[24] M. A. Arrio, P. Sainctavit, C. Brouder, and C. Deudon, XMCD sum rules at Cu2+ L3,3 edge, Physica B 208, 27 (1995).
[25] M. Retegan, Crispymat: v0.7.3 (2019), doi: 10.5281/zenodo.1008184.
[26] M. W. Haverkort, M. Zwillerzycki, and O. K. Andersen, Multiplet ligand-field theory using Wannier orbitals, Phys. Rev. B 85, 165113 (2012).
[27] S. Blundell, Magnetism in condensed matter, *Oxford Master Series in Condensed Matter* (Oxford University Press Inc., New York, 2001).

[28] P. Audehm, M. Schmidt, S. Bruck, T. Tietze, J. Grafe, S. Macke, G. Schutz, and E. Goering, Pinned orbital moments- A new contribution to magnetic anisotropy, *Sci. Rep.* 6, 25517 (2016).

[29] I. G. Rau, S. Baumann, S. Rusponi, F. Donati, S. Stepanow, L. Gragnaniello, J. Dreiser, C. Piamonteze, F. Nolting, S. Gangopadhyay, O. R. Albertini, R. M. Macfarlane, C. P. Lutz, B. A. Jones, P. Gambardella, A. J. Heinrich, and H. Brune, *Science* 344, 988 (2014).

[30] M. Salluzzo et al., Origin of Interface Magnetism in BiMnO3/SrTiO3 and LaAlO3/SrTiO3 Heterostructures, *Phys. Rev. Lett.* 111, 087204 (2013).

[31] J. S. Lee, Y. Xie, H. Sato, C. Bell, Y. Hikita, H. Hwang, and C. Kao, Titanium dxy ferromagnetism at the LaAlO3/SrTiO3 interface, *Nat. Mater.* 12, 703 (2013).

[32] F. Gunkel, D. V. Christensen, Y. Z. Chen, and N. Pryds, Oxygen vacancies: The (in)visible friend of oxide electronics, *Appl. Phys. Lett.* 116, 120505 (2020).

[33] J. R. L. Mardegan, D. V. Christensen, Y. Z. Chen, S. Parchenko, S. R. V. Avula, N. Ortiz Hernandez, M. Decker, C. Piamonteze, N. Pryds, and U. Staub, Magnetic and electronic properties at the γ-Al2O3/SrTiO3 interface, *Phys. Rev. B* 99, 134423 (2019).

[34] P. Gambardella et al., Giant magnetic anisotropy of single cobalt atoms and nanoparticle, *Science* 300, 1130 (2003).

[35] C. Piamonteze, P. Miedema, and F. M. F. de Groot, Accuracy of the spin sum rule in XMCD for the transition-metal L edges from manganese to copper, *Phys. Rev. B* 80, 184410 (2009).

[36] J. A. Krieger et al., Do topology and ferromagnetism cooperate at the EuS/Bi2Se3 interface? *Phys. Rev. B* 99, 064423 (2019).