Bidirectional ROMP of paracyclophane-1,9-dienes to tri- and penta-block p-phenylenevinylene copolymers†

Venukrishnan Komanduri, Dharam R. Kumar, Daniel J. Tate, Raymundo Marcial-Hernandez, Benjamin J. Lidster and Michael L. Turner

Dialkoxy and dialkyl substituted paracyclophane-1,9-dienes undergo bidirectional ring opening metathesis polymerisation (ROMP) on addition of bifunctional Hoveyda–Grubbs initiators. The living nature of the polymerisation was demonstrated for different [monomer]/[initiator] ratios and block copolymers were prepared by sequential ROMP and ROMP-ATRP to give fully conjugated rod–rod–rod and partially conjugated coil–rod–coil ABA and BAB triblock copolymers, respectively. These strategies were successfully extended to the synthesis of the corresponding pentablock copolymers. Optical and electrochemical properties of cis/trans and all trans homo and block copolymers are discussed.

Introduction

π-Conjugated polymers have found extensive application as active layers in the fabrication of optoelectronic and electronic devices such as OLEDs, OFETs, OPVs, and sensors. Key advantages for the use of polymeric materials include improved flexibility, mechanical robustness, lighter weight and ease of processing. The performance of these devices is greatly influenced by the morphology of the active layer, and this is strongly influenced by the kinetics of thin film formation during device fabrication. An attractive approach to obtaining reproducible thin film morphologies is to utilise the thermodynamically driven phase separation of block copolymers, as these materials form well defined nanostructures that can be self-assembled in biphasic media. Block copolymers with conjugated segments can either be fully conjugated rod–rod systems, in a diblock or rod–coil systems where the conjugated block is coupled to a saturated polymer backbone. There are only a few polymerisation methods that give well-defined fully conjugated block copolymers. These include Kumada catalyst-transfer polymerisation (KCTP also known as GRIM), a nickel catalysed Negishi type polymerisation and a palladium catalyst transfer Suzuki–Miyaura polycondensation. The synthesis of rod–coil systems has been more extensively reported, including the ROMP of cyclooctadiene followed by ATRP of the resulting macroinitiator. Phenylenevinylene (PPV) block copolymers can be prepared by the Siegrist polycondensation or by complex multistep syntheses. However these approaches lead to poor control of the molecular weight distribution and result in polymers with broad polydispersities (Đ) and significant backbone defects.

Ring opening metathesis polymerisation (ROMP) of strained paracyclophanedienes and cyclopentadienes gives well-defined PPV homopolymers and fully conjugated diblock copolymers. Rod-coil PPV diblock copolymers can be prepared by the sequential one-pot ROMP of tetraalkoxy paracyclophanediene and oxanorbornene monomers. This sequential ROMP strategy was adopted by Weck et al. using di, tetraalkoxy cyclophanedienes and norbornene ester monomers to access PPV tri and tetrablock copolymers. A step-wise strategy using monotelechelic PPV blocks as macroinitiators for ATRP of methylmethacrylate gives rod-coil PPV-PMMA diblock copolymers. Rod–coil PPV block copolymers have also been prepared through the preparation of PPV macroinitiators using anionic polymerisation followed by the introduction of the coil unit using a radical polymerisation. However, to date there have been very few reports on the preparation of fully and partially conjugated PPV triblock copolymers. These include an ADMET polymerisation/Wittig olefination strategy to prepare fully conjugated ABA and ABCBA type PPV block copolymers and the synthesis of donor–acceptor–donor triblock copolymers by a Knoevenagel condensation followed by a Nickel catalysed Yamamoto coupling by Tu et al. Coir–rod–coil ABA triblock copolymers with pendant PEG chains have been prepared by a Heck coupling...
Results and discussion

Initial optimisation studies focussed on the ROMP of monomers M1 and M2 using the bifunctional catalyst 1. The required cyclopane-1,9-diene monomers M1 (R = O-2-ethylhexyl) and M2 (R = n-octyl) were prepared using standard synthetic procedures. The bifunctional catalyst 1 was synthesised by cross metathesis of Grubbs second generation catalyst G2 with 2,5-dimethoxy-1,4-divinylbenzene as reported.

The first ROMP experiment was performed using dialkoxy monomer M1 ([M1]/[I] = 10) in DCM-d2 at room temperature. The polymerisation was followed by in situ 1H NMR spectroscopy and complete monomer consumption was observed after 63 h. However, reaction times were reduced to 10 h when the ROMP was performed at 60 °C in TCE-d2 (Fig. S3†). After consumption of the monomer the polymerisation was quenched with excess ethylvinylether and stirred at rt for 4 h. Precipitation of the crude product (MeOH/Celite), followed by elution with chloroform gave telechelic PPV 2a (Mn(calc.) = 4.7 kDa, Mn(obs.) = 5.7 kDa, Dm = 1.40) in 95% yield (entries 1 and 2, Table S1†).

Bidirectional ROMP of the dialkyl monomer M2 ([M2]/[I] = 10) in DCM-d2 at room temperature was complete in 1.5 h but gave poor control over the molecular weight distribution (Mn(calc.) = 4.4 kDa, Mn(obs.) = 79 kDa, Dm = 1.8) due to incomplete initiation and fast propagation. The higher propagation rates are due to the absence of Lewis basic sites in monomer M2 that can ligate to the intermediate 14 electron ruthenium complex formed after initiation. Therefore, the ROMP was performed in the presence of an external ligand 3-BrPy (40 mol%) at 40 °C in TCE-d2 for 20 h (Fig. S4†) to slow down propagation. This gave PPV 3a with significant control over molecular weight (Mn(obs.) ~ 7.1 kDa vs. 79 kDa in the absence of 3-BrPy, entries 3 and 4, Table S1†). Further control over the molecular weight and an improved yield was obtained (Mn(obs.) ~ 5.4 kDa) when the ROMP was performed at 40 °C using anhydrous degassed DCM as solvent (entries 4 and 5, Table S2†). This could be due to the better solubility of 1 in DCM. The living nature of the polymerisation was demonstrated by varying the [monomer]/[initiator] ratio to obtain polymers 2a–c (96–93% yield) and 3a–c (95–81% yield) as shown in Fig. 2 (also, see Table S2†). The [M]/[I] ratios given for the monomers M1 and M2 are different because ROMP of M1 under the optimal conditions gives poor control of the molecular weight and dispersity beyond ratios of 20 : 1.

A powerful application of this bidirectional ROMP methodology was demonstrated by the synthesis of both rod–rod–rod and coil–rod–coil triblock copolymers. Triblock conjugated copolymers have found interesting application as the active layer in OPV and OLED devices but only limited synthetic
methods have been reported. AB and ABA block copolymers with PPV units are known to self-assemble into nanostructures, the morphology of which depends on the mole fraction of the individual blocks.\textsuperscript{6,43}

The α-bromoester end capped PPV macroinitiators 4a and 5a were prepared in excellent yields (95–98\%) by bidirectional ROMP of monomers M1 and M2 followed by quenching with α-bromoester phenyl methacrylate (MMA) in hot toluene for 3 h. Aliquots of the reactions were terminated at different time intervals (30, 90, 180 min) by exposing to air followed by precipitation into excess methanol. The solids were redissolved in chloroform and precipitated by addition of diethyl ether. Upon filtration, PMMA-b-PPV-b-PMMA triblock polymers 6a-c and 7a-c were obtained as orange/yellow solids with dispersities ranging from 1.23 to 1.33 and a unimodal molar mass distribution by GPC analysis (Scheme 1, Fig. S21 and Table S3\textsuperscript{†}).

These macroinitiators (4a and 5a) were subjected to CuBr mediated atom transfer radical polymerisation (ATRP)\textsuperscript{33} with methyl methacrylate (MMA) in hot toluene for 3 h. Aliquots of the reactions were terminated at different time intervals (30, 90, 180 min) by exposing to air followed by precipitation into excess methanol. The sols were redissolved in chloroform and precipitated by addition of diethyl ether. Upon filtration, PMMA-b-PPV-b-PMMA triblock polymers 6a-c and 7a-c were obtained as orange/yellow solids with dispersivities ranging from 1.23 to 1.33 and a unimodal molar mass distribution by GPC analysis (Scheme 1, Fig. S21 and Table S3\textsuperscript{†}).

The MMA content in these triblock copolymers was determined from the \textsuperscript{1}H NMR spectra by a comparison of the integration of the OMe group of MMA (δ = 3.98–3.17 ppm) against the combined phenyl and vinylene protons for 6a-c (δ = 7.93–6.34 ppm) and methylene protons for 7a-c (δ = 2.86–2.16 ppm) respectively (Fig. S22–27, Table S3\textsuperscript{†}).

Bidirectional living polymerisation of cyclophanedienes in a sequential manner gave fully conjugated triblock copolymers. ROMP of monomer M1 with initiator 1 ([M1]/[1] = 10) in TCE at 60 °C was monitored by GPC and TLC analyses (Fig. S47\textsuperscript{†}). After complete consumption (t = 8 h) of monomer M1 an external ligand, 3-BrPy (40 mol%), and a solution of the second monomer M2 in TCE were added. The reaction mixture was stirred at 40 °C until the second monomer M2 was consumed (t = 12, 16 and 18 h) and the polymerisation was quenched with excess ethylvinylether. Purification by precipitation (Celite/MeOH) and elution with chloroform afforded fully conjugated ABA triblock polymers 8a-c in 80–88\% yields with dispersivities ranging from 1.87–1.97. The higher polydispersities and the shoulder peak observed in the GPC traces of block copolymer 8c can be assigned to partial isomerisation of the cis-alkenes (Scheme 2, left and Table S4\textsuperscript{†}). The sequential ROMP strategy was more efficient (yields 93–97\% and \(D_m = 1.35–1.87\)) when the monomer addition sequence was reversed to give BAB triblock copolymers 9a-c containing alkyl PPV segments at the center and alkoxy PPV units on either side (Fig. S48, Scheme 2, right and Table S4\textsuperscript{†}).

The triblock copolymers 8a-c and 9a-c were characterised by \textsuperscript{1}H NMR spectroscopy and the ratio of monomer units M1 and M2 incorporated was calculated by combined integration of the alkoxy methylene protons corresponding to M1 and the methoxy protons of 1 (δ 4.12–3.77 ppm) against the benzyl

\[\begin{align*}
\text{Scheme 1} & \quad \text{ROMP-ATRP strategy to prepare coil–rod–coil ABA phenylethenylene triblock copolymers 6a–c and 7a–c (GPC in THF).}
\end{align*}\]

\[\begin{align*}
\text{Scheme 2} & \quad \text{Sequential ROMP strategy to fully conjugated rod–rod–rod ABA (8a–c) and BAB (9a–c) phenylethenylene triblock copolymers (GPC in THF).}
\end{align*}\]
protons corresponding to \(M_2\) (δ 2.72–2.20 ppm) (Fig. S31–41†). The individual block content ratios determined from the \(^1\)H NMR spectra were in good correlation with the calculated values based on the monomer to initiator ratio (Table S4†).

\(^1\)H NMR spectral analysis of the BAB triblock copolymers \(9\text{a-c}\) (Fig. S37†) indicated high vinyl end-capping efficiency (\(f > 80\%\)) suggesting that further functionalisation/elaboration could be possible. Therefore the sequential ROMP was quenched with the \(\alpha\)-bromooester substituted phenylmethyl ether \(E\) instead of ethyl vinyl ether to give the bifunctional triblock PPV macroinitiator \(10\) in excellent yield with excellent end capping efficiency (95% yield, \(f > 87\%\)). Under similar ATRP reaction conditions to those in Scheme 1 the macroinitiator \(10\) and MMA gave coil–rod–rod–rod–coil CBABC pentablock copolymers \(11\text{a–c}\) with molecular weights ranging from 20.6–36.5 kDa and dispersities in the range 1.49–1.25. The slightly broader dispersity of \(11\text{a}\) is due to partial isomerisation of the cis-alkenes (Scheme 3, Table S3 and Fig. S28–30†).

Furthermore the sequential ROMP approach can be used to synthesise fully conjugated pentablock copolymers as shown in Scheme 4. Dialkyl monomer \(M_2\) was subjected to ROMP ([\(M_2]\]/[1] = 10) in the presence of 3-BrPy ligand and upon complete consumption the dialkoxy monomer \(M_1\) was added and stirred for 15 h, complete consumption of this monomer (\(M_1\)) was followed by addition of either \(M_2\) or a mixture of the isomeric tetraalkyl monomers \(M_3, M_4\) (\(M_4\) being unreactive). The polymerisation reaction was eventually quenched by adding excess ethylvinylnether and the fully conjugated ABABA and CBABC type pentablock copolymers \(12\) and \(13\) were isolated by precipitation (Scheme 4, S43 & S45†).

Dilute solutions of conjugated polymers \(2\text{a–c}, 3\text{a–c}, 8\text{a–c}, 9\text{a–c}\), \(12\) and \(13\) in degassed anhydrous dichloromethane were subjected to photoisomerisation under UV irradiation at 365 nm for 48–72 h. \(^1\)H NMR analysis of the polymers obtained indicated complete isomerisation and an all trans geometry of the backbone vinylenes (Fig. 4a, also see ESI†). In the GPC a lower retention time was observed for the all trans isomers as expected because of the higher hydrodynamic volume of this polymer (Fig. 4b).

The optical properties of the parent homopolymers were measured in solution (Fig. S49 and Table S5†) and the block copolymers were recorded in both solution and as thin films (Fig. 5, S50–54, and Table S5†). The absorption spectra of the fully conjugated tri and pentablock copolymers \(8\text{a–c}, 9\text{a–c}, 12\) and \(13\) showed \(\lambda_{\text{max}}\) values around 450 nm and 380 nm correlating to absorptions of the individual blocks \(2\text{a–c}\) and \(3\text{a–c}\). Increasing the content of monomer \(M_2\) in polymers \(8\text{a to 8c}\) increased the intensity at 370 nm in direct correlation with the ratio of the individual blocks (Fig. 5a). For polymers \(9\text{a–c}\) the absorption maxima were red shifted going from \(9\text{a} to 9\text{c} with Stock Image
increasing alkoxy monomer content (Fig. 5b). Excitation of the block copolymers 8a-c, 9a-c, 12 and 13 at 370 nm and the corresponding all trans isomers at 440 nm resulted in emission maxima approximately at 520 nm (Table S5†). The absorption and emission spectra of rod-coil block copolymers 6a-c, 7a-c and 11a-c showed absorption maxima at 460 nm (6a-c), 400 nm (7a-c), 370 nm (11a-c) and emission maxima at 527 nm (6a-c), 491 nm (7a-c) and 520 nm (11a-c). These values agree well with those of parent macroinitiators 4a, 5a and 10, indicating that the PMMA block has no influence on the optical properties of these materials. The $\Phi_{\text{fl}}$ of the macroinitiators 4a, 5a and 10 and the block copolymers 7a-c, 8a and 11a-c are identical and suggests that the radical polymerisation did not perturb the $\pi$-conjugated backbone (Table S5†). A small red shift in emission maxima values (≈20 nm) was observed in the solid state spectra of the block copolymers consistent with previous studies on PPV polymers (Fig. S53–54 and Table S6†).28

The electrochemical properties of the homopolymers (2a-c and 3a-c) and fully conjugated block copolymers (8a-c, 9a-c, 12 and 13) were studied by cyclic voltammetry in dichloromethane using [C$_6$H$_5$]$_2$NH$_2$PF$_6$ (0.1 M solution in DCM) as the electrolyte in conjunction with a platinum working electrode. The calculated HOMO and LUMO energy levels for 2a-c are consistent with the reported values (HOMO = −5.14 and LUMO = −2.56 eV) for these polymers.28 The homopolymers 3a-c have deeper HOMO levels (≈−5.36 eV) (Table 1, entries 1, 2, 3 and Table S7†).

The HOMO and LUMO energy levels of the tri- and pentablock copolymers are consistent with those of the parent homopolymers. The cis/trans ABA triblock BCPs 8a-c exhibited reversible oxidation $E_{1/2}$ values in the range of 0.6–0.4 V, with slight increase in HOMO energy levels (−5.23 to −5.05 eV) but the BCPs 9a-c showed the same HOMO value (≈−5.14 eV) indicating localisation of the HOMO on the dialkoxyphenylene vinylene block in 9a-c. The LUMO energy levels for BCPs 8a-c and 9a-c are in a similar range, −2.69 to −2.78 eV for 8a-c and −2.60 to −2.67 eV for 9a-c (Table S7† and Table 1, entries 5–7 & 9–11). The cis/trans pentablock copolymers 12 and 13 showed reversible oxidation and reduction potentials $E_{1/2}$ at 0.5–0.6 V and −1.99, −1.80 V with the HOMO energy level approximately −5.14 eV but slightly deeper LUMO level (−2.79 eV) for BCP 13 (Table S7† and Table 1, entries 13 & 15). In general, the all trans homo and block copolymers have a higher lying HOMO when compared to their cis/trans forms due to extended conjugation in the trans form. Also, a consistent deepening of LUMO levels was observed for all the trans polymers except for trans-13 (Table 1, entries 15 & 16). Finally, a clear trend in bandgap reduction was observed for the all trans polymers due to extended delocalisation and both the electrochemical and optical bandgap values for all the polymers are comparable (Table 1).

### Table 1: Electrochemical properties of homopolymers (2a & 3a) and block copolymers (8a-c, 9a-c, 12 & 13)

| Entry | Polymer | HOMO $^a$ (eV) | LUMO $^b$ (eV) | $E_{\text{g (elc)}}$ $^d$ (eV) |
|-------|---------|----------------|----------------|------------------|
| 1     | 2a      | −5.14          | −2.66          | 2.48 (2.30)      |
| 2     | 2a (E)  | −5.01          | −2.84          | 2.17 (2.28)      |
| 3     | 3a      | −5.36          | −2.58          | 2.78 (2.50)      |
| 4     | 3a (E)  | −5.10          | −2.81          | 2.29 (2.46)      |
| 5     | 8a      | −5.23          | −2.77          | 2.46 (2.31)      |
| 6     | 8b      | −5.12          | −2.69          | 2.43 (2.27)      |
| 7     | 8c      | −5.05          | −2.78          | 2.27 (2.27)      |
| 8     | 8b (E)  | −5.03          | −2.70          | 2.33 (2.29)      |
| 9     | 9a      | −5.10          | −2.67          | 2.43 (2.33)      |
| 10    | 9b      | −5.14          | −2.66          | 2.48 (2.32)      |
| 11    | 9c      | −5.14          | −2.60          | 2.54 (2.30)      |
| 12    | 9a (E)  | −5.02          | −2.75          | 2.27 (2.30)      |
| 13    | 12      | −5.14          | −2.64          | 2.50 (2.34)      |
| 14    | 12 (E)  | −5.04          | −2.69          | 2.35 (2.29)      |
| 15    | 13      | −5.19          | −2.79          | 2.40 (2.31)      |
| 16    | 13 (E)  | −5.07          | −2.65          | 2.38 (2.28)      |

$^a$ HOMO = ($\Phi_{\text{ox}}$ + $F_{\text{cH}}$) + 4.8, $^b$ LUMO = ($\Phi_{\text{red}}$ − $F_{\text{ox}}$) + 4.8, $^c$ $E_{\text{g (elc)}}$ = HOMO−LUMO. $^d$ Optical band gap values in parenthesis ($E_{\text{g (opt.)}}$ = 1240/λ$_{\text{onset}}$).

### Conclusions

In summary, the first bidirectional ROMP of strained paracyclophane-1,9-dienes using a bifunctional Hoveyda-Grubbs initiator is reported. This methodology was used in the controlled synthesis of rod–rod and rod–coil tri- and pentablock copolymers through sequential ROMP and ROMP-ATRP reactions. The initially formed cis,trans polymers can be photoisomerised to the corresponding all trans forms. The optical properties of the block copolymers directly correlate with those of the individual polymer blocks and complete energy transfer between blocks is observed in the fluorescence emission spectra of thin films. Fluorescence quantum efficiency measurements of the rod–coil block copolymers showed no damage to $\pi$-conjugated backbone under radical ATRP reaction conditions.

### Conflicts of interest

There are no conflicts to declare.

### Acknowledgements

The authors would like to thank the European Commission for the award of a Marie-Curie International Fellowship (CyclAr) to VK, the Indian High Commission Education Wing, London for PhD studentship to DRK (11015/16/2012-SCD-V) and the EPSRC for a studentship award to BJL. The authors would also like to thank the EPSRC for funding of the NMR spectrometers under grant EP/K039547/1 and Dr Ralph Adams, Mr Carlo Bawn for technical advice and Mr Adam Woodward for his technical support in measuring fluorescence quantum yields.
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