Effects of strongly selective additives on volume phase transition in gels
Yuki Uematsu and Takeaki Araki
Department of Physics, Kyoto University, Kyoto 606-8502, Japan

(Dated: 1 May 2014)

We investigate volume phase transition in gels immersed in mixture solvents, on the basis of a three-component Flory-Rehner theory. When the selectivity of the minority solvent component to the polymer network is strong, the gel tends to shrink with an increasing concentration of the additive, regardless of whether it is good or poor. This behavior originates from the difference of the additive concentration between inside and outside the gel. We also found the gap of the gel volume at the transition point can be controlled by adding the strongly selective solutes. By dissolving a strongly poor additive, for instance, the discontinuous volume phase transition can be extinguished. Furthermore, we observed that another volume phase transition occurs far from the original transition point. These behaviors can be well explained by a simplified theory neglecting the non-linearity of the additive concentration.

I. INTRODUCTION

Swelling behavior of a polymer network was first investigated by Flory and Rehner. On the basis of this work, discontinuous volume phase transition (VPT) of gels was predicted theoretically. Thence, the VPT has received much attention from both scientific and industrial viewpoints. In various types of gels, addition of solutes into the solvent often plays very important roles in their volume change. For example, the VPT was first realized in a polyacrylamide (PAA) gel immersed in a mixture of water and acetone. In this experiment, the solvent quality, or \( \chi \) parameter, is controlled by changing the volume fraction of the acetone. It is also well known that addition of salts dramatically affects the behaviors of ionic gels. Furthermore, a nonionic hydrated gel such as poly(\( N \)-isopropylacrylamide) gel (NIPA) changes its volume discontinuously. In the hydrated gels, the VPT is attributed to temperature-dependence of hydrogen bondings of the polymer network. It has been reported that its VPT is affected by adding salt and other additives.

It is often assumed that the composition of a mixture solvent is constant in and out of a gel. This assumption is referred to as single liquid approximation (SLA). The SLA is experimentally confirmed in a NIPA gel in a dimethyl sulfoxide-water mixture, for instance. However, the compositions of mixture solvents are not necessarily homogeneous. There, the difference of the composition between the interior and exterior of the gel would lead to a dramatic effect on its volume change. Some extended Flory-Huggins models dealing with a polymer network and two species of solvent molecules were developed to investigate the volume change of a polymer network in a mixture solvent. In these theories, the volume change is characterized by the compositions of the three components and three \( \chi \) parameters. They were studied systematically for fixed sets of the \( \chi \) parameters. Usually, the \( \chi \) parameters in polymer systems depend on environment parameters such as temperature. Since it is quite complicated to incorporate the temperature dependences of the \( \chi \) parameters, the swelling behavior in mixture solvents has not been fully understood.

Recently, the effects of solutes with strong selective solvation on phase behaviors of a water-oil mixture was investigated. By dissolving a very small amount of strongly hydrophobic solute into the water-oil mixture, the oil-rich phase containing the solute is precipitated even in the one-phase region. This finding implies that such solutes would affect the VPT in gels.

The aim of this study is to clarify the effect of the additives on the VPT in gels, based on a simple theoretical argument. In particular, we focus on the cases, in which one of the solvent components has strong selectivity to the polymer network.

II. MODEL FREE ENERGY

Swelling behaviors of gels are well described by the Flory-Rehner theory. The free energy \( F \) consists of a mixing part \( F_{\text{mix}} \) and an elastic part \( F_{\text{el}} \) as

\[
F = F_{\text{mix}} + F_{\text{el}}. \tag{1}
\]

For a gel immersed in a mixture solvent, the mixing free energy \( F_{\text{mix}} \) is given by

\[
F_{\text{mix}} = k_B T v_0^{-1} \left[ V_g f(\phi_{1g}, \phi_{2g}, \phi_3) + V_s f(\phi_{1s}, \phi_{2s}, 0) \right], \tag{2}
\]

where \( k_B \) is the Boltzmann constant and \( T \) is the temperature. \( V_g \) and \( V_s \) are the volumes of the interior and exterior of the gel, respectively. In this model, we assume that the volumes of each monomer of the polymer network and the solvent molecules are equal to a characteristic volume \( v_0 \) for simplicity. \( f(\phi_1, \phi_2, \phi_3) \) is the Flory-Huggins type mixing free energy of the ternary system as

\[
f(\phi_1, \phi_2, \phi_3) = \phi_1 \ln \phi_1 + \phi_2 \ln \phi_2 + \sum_{i<j} \chi_{ij} \phi_i \phi_j. \tag{3}
\]

Here, \( \phi_1 \) and \( \phi_2 \) are the volume fractions of the first and second components of the mixture solvent, respectively.
In Eq. (2), \( \phi_{i g} \) and \( \phi_{i a} \) stand for the volume fraction of the \( i \)-th component \((i = 1, 2)\) in and out of the gel. \( \phi_{3} \) is the volume fraction of the polymer network. Since all the polymer chains are chemically connected forming a single network, the polymers are not dissolved outside the gel, that is, \( \phi_{3b} = 0 \). In Eq. (3), the first two terms stem from the translational entropy of the solvent molecules. Here, the translational entropy of the network is neglected. The last term is the interaction energy among the solvent molecules and the monomers of the network. \( \chi_{ij} \) is the interaction parameter between the \( i \)-th and \( j \)-th components.

We employ Flory’s rubber elasticity as\(^{32}\),

\[
F_{el} = \frac{1}{2} k_B T \nu V_{g0} \left\{ \sum_{k=x,y,z} \gamma_k^2 - 3 - 2B \ln(\gamma_x \gamma_y \gamma_z) \right\},
\]

where \( V_{g0} \) is the volume of the gel in a reference state, \( \nu \) is the density of cross linking points, and \( \gamma_k \) is the elongation ratio in the \( k \)-axis \((k = x, y, z)\). \( B \) is a nonlinear elastic constant, which is often assumed \( B = 1/2 \).

In polyelectrolyte gels, the translational entropy of the elongation ratio is coupled with the volume fraction \( \phi_{3} \) explicitly. In the case of isotropic swelling, the elongation ratio in the \( k \)-axis is the density of cross linking points, and \( \gamma_k \) is the elongation ratio in the \( k \)-axis \((k = x, y, z)\). \( B \) is a nonlinear elastic constant, which is often assumed \( B = 1/2 \).

In this study, we assume that the binary solvent is given by

\[
\Pi_0 = -\phi_3 - \chi_{31} \phi_3^2 - \ln(1 - \phi_3).
\]

If the solvent does not contain the additive \( \phi_2 = 0 \), the balance of the osmotic pressure is expressed by \( \Pi_0 = \Pi_{el} = 0 \).

Here, we define a grand potential as

\[
\Omega = F_{mix} + F_{el} - \mu_2 (V_g \phi_{2g} + V_s \phi_{2s}) + \kappa (V_g + V_s),
\]

where \( \mu_2 \) and \( \kappa \) are Lagrange multipliers to conserve the amount of the second component and the total volume. The equilibrium state is characterized by minimizing \( \Omega \),

\[
\frac{\partial \Omega}{\partial \phi_{2g}} = \frac{\partial \Omega}{\partial \phi_{2s}} = 0, \quad \frac{\partial \Omega}{\partial V_g} = \frac{\partial \Omega}{\partial V_s} = 0.
\]
which the gel can undergo VPT without additives. We set $\phi_{30} = 5.0 \times 10^{-2}$, $\nu v_0 = 1.0 \times 10^{-2}$ and $B = 0.75$. We vary $\chi_{31}$ continuously to induce the volume change with fixing the other parameters. We obtain the swelling curves of the gel by solving Eqs. (12) and (13) numerically. Figure 1 shows the swelling curves for several values of $\phi_{2g}$. In the absence of additives, the VPT occurs at $\chi_{31} = \chi^{(0)}_{31} (\approx 0.93)$. In Fig. 1 (a), we dissolve the additive of $\chi_{12} = 0.0$ and $\chi_{23} = -1.0$, which has affinity to the polymer network (pro-gel). As its concentration $\phi_{2g}$ is increased, the transition point shifts to higher $\chi_{31}$. In Fig. 1 (b), the additive of $\chi_{12} = 0.0$ and $\chi_{23} = 2.0$ is dissolved into the solvent. This additive dislikes the polymer network (anti-gel). It is shown that the transition point shifts to lower $\chi_{31}$ with an increasing $\phi_{2g}$. For the cases as in Fig. 1, the SLA can explain the effects of the solute on the volume changes qualitatively. In the SLA, the volume fractions of the solvent components inside the gel are assumed to be $\phi_{1g} = (1 - \phi_3)(1 - \phi_{2g})$ and $\phi_{2g} = (1 - \phi_3)\phi_{2g}$. Then, the interaction parameter between the polymer network and the mixture solvent is approximated as

$$\tilde{\chi}_{31} = \chi_{31}(1 - \phi_{2g}) + \chi_{23}\phi_{2g} - \chi_{12}\phi_{2g}(1 - \phi_{2g}) = \chi_{31} + G\phi_{2g} + O(\phi_{2g}^2). \quad (18)$$

This means that the interaction parameter changes effectively with $\phi_{2g}$ depending on $G$. For the plotted range in Fig. 1 (a) ($\chi_{31} \in [0.92, 0.95]$), $G$ remains negative, so that the solvent becomes more good, swelling the gel with an increasing $\phi_{2g}$. As shown in Fig. 1 (b), on the other hand, the solvent changes to more poor and the gel shrinks with $\phi_{2g}$ for the solute of positive $G$.

In Fig. 1, the absolute value of the resulting $G$ is rather small ($|G| < 2$). Next, we study the effects of solutes of strong selectivity $|G| \gg 1$. Figures 2 (a) and (b) show the swelling curves in the mixture solvents of $\chi_{23} = -1.0$ and $\chi_{23} = 1.10$, respectively. The other parameters are the same as those in Fig. 1. In the both cases, the transition points are shifted to lower $\chi_{31}$, regardless of whether the solute is good or poor. This behavior is in contrast to the volume changes in Fig. 1 and indicates the SLA does not work well when the additives have the strong selectivity to the polymer network.

Figure 2 (a) shows the dependences of the transition shift $\Delta \chi_{31}^{(0)}$ on the additive concentration $\phi_{2g}$, where $\Delta \chi_{31}^{(0)} = \chi_{31}^{(0)} - \chi_{31}^{(0)}$ and $\chi_{31}^{(0)}$ is the transition point without additives. An increase of $\Delta \chi_{31}^{(0)}$ is observed for $\chi_{23} = -1.0$, whereas $\Delta \chi_{31}^{(0)}$ is lowered with an increasing $\phi_{2g}$ for the other additives. In the plotted range of $\phi_{2g}$, $\Delta \chi_{31}^{(0)}$ has linear dependence on $\phi_{2g}$. The transition shifts for $\phi_{2g} = 0.01$ are plotted with respect to $\chi_{23}$ in Fig. 3 (b). It indicates a non-monotonic behavior of the transition point.

The difference of the additive effects between $\chi_{23} = -1.0$ and $\chi_{23} = -9.0$ is notable, since the both additives have tendencies to be adsorbed onto the polymer network. When $|\chi_{23}|$ is small, the difference between $\phi_{2g}$ and $\phi_{2a}$ is rather small. As explained by Eq. (18), the addition of the solute with negative $\chi_{23}$ makes the mix-
volume gap is increased for $\chi_{23}$ is large enough, on the other hand, the mixture solvent inside the gel becomes much denser than that outside the gel. By shrinking its volume, the polymer network tends to in-

ture solvent more good to the polymer. Thus, the gel is simply swollen with an increasing $\phi_{2s}$. If $|\chi_{23}|$ is large enough, on the other hand, the mixture solvent inside the gel does not undergo the VPT [see below]. Figure 4 plots the swelling curve becomes non-monotonic with respect to $\chi_{23}$ = -9.0. The black solid lines are calculated from Eq. (21)

![Figure 4](image1.png)

FIG. 4. The gap of the volume fraction $\phi_3$ at the transition is plotted with $\phi_{2s}$. We set $\phi_{30} = 5.0 \times 10^{-2}$, $\nu v_0 = 1.0 \times 10^{-2}$, $B = 0.75$, and $\chi_{12} = 0.0$ as in Figs. 1 and 2.

Substituting Eq. (19) into Eq. (17), we obtain

$$\Pi_{ad} = \ln(1-\phi_{2s}) - \ln(1-\phi_{2s} e^{-G\phi_3}) + G\phi_3(1-\phi_3) e^{-G\phi_3} + O(\phi_{2s}^2)$$

$$= [e^{-G\phi_3} \{1 + G\phi_3(1-\phi_3)\} - 1] \phi_{2s} + O(\phi_{2s}^2).$$

Then, the swelling curve is approximately given from Eqs. (19), (19) and (20) by

$$\chi_{31} \equiv -\phi_3^{-2} \ln(1-\phi_3) - \alpha\phi_3^{-5/3} + (\beta - 1)\phi_3^{-1}$$

$$+ [e^{-G\phi_3} \{1 + G\phi_3(1-\phi_3)\} - 1] \phi_3^{-2} \phi_{2s},$$

where $\alpha = \nu v_0 / \phi_{30}^{1/3}$ and $\beta = \nu v_0 B / \phi_{30}$. Since the right hand side of Eq. (21) includes $\chi_{31}$ via $G$, we cannot obtain an analytical solution of $\chi_{31}$. Then, we replace $G$ to $G = \chi_{23} - \chi_{31t} - \chi_{23}$ and regard $G$ as a fixed parameter. The approximated curves of Eq. (21) are drawn in Figs. 1, 2 and 5. They are well in agreement with the numerical solutions.

Regarding $\beta$ as a variable, a tri-critical point of the VPT is given by

$$\frac{\partial \chi_{31}}{\partial \phi_3} = 0, \quad \frac{\partial^2 \chi_{31}}{\partial \phi_3^2} \phi_{3c}, \beta_c = 0$$

Here, $\alpha$ is treated to be fixed. By solving Eq. (22), we obtain $\phi_{3c}$ and $\beta_c$, which give the swelling curve passing through the tri-critical point. In the vicinity of the tri-critical point, the swelling curve is approximated by

$$\chi_{31} \approx \chi_{31t} - \beta - \beta_c (\phi_3 - \phi_{3c}) + \frac{u}{6}(\phi_3 - \phi_{3c})^3,$$

where $u$ is a positive constant. If $\beta < \beta_c$, the curve of $\chi_{31}$ changes monotonically with $\phi_3$, such that the gel volume changes continuously with changing $\chi_{31}$. On the other hand, inflection points appears when $\beta > \beta_c$. If so, the gel becomes mechanically unstable, so that it exhibits the discontinuous volume changes as $\Delta \phi_{31} = 2\sqrt{(\beta - \beta_c)/u} \phi_{3c}$ at $\chi_{31} = \chi_{31c}$.

The part of the osmotic pressure $\tilde{\Pi}_{ad}$ shifts the tri-critical point in Eq. (23) as

$$\chi_{31} = \chi_{31}^{(0)} + g(\phi_{3c}, G) \phi_{2s}.$$

$$\beta_c = \beta_c^{(0)} + h(\phi_{3c}, G) \phi_{2s}^2.$$
even in a gel, which originally shows a continuous volume phase transition point is increased linearly with $\phi_{2s}$ in the region of for positive $g$. In the region of negative $h$, the volume gap of the transition point is increased.

Here, $\phi_{3c}^{(0)}$ gives the tri-critical point in the case of $\phi_{2s} = 0$. From Eq. (21), the prefactors $g$ and $h$ are given by

\begin{align}
  g(\phi_3, G) &= \phi_3^{-2}[e^{-G\phi_3}(1 + G\phi_3(1 - \phi_3)) - 1], \\
  h(\phi_3, G) &= \phi_3^{-3}[2 - e^{-G\phi_3}G^2\phi_3^2(1 - \phi_3) + 2G\phi_3 + 2].
\end{align}

$g$ and $h$ change their signs depending on $\phi_3$ and $G$. Their dependences are shown in Fig. 6. If $g(\phi_3, G)$ is positive, the transition point shifts to higher $\chi_{31}$ with an increasing $\phi_{2s}$ and vice versa. Eq. (26) for $\phi_{2s} = 0.01$ is drawn in the broken curve in Fig. 3 (b). Here, we set $\phi_{3c} = 0.2392$ and $\chi_{31} = 0.93$. This approximated curve is in agreement with the numerical solution. When $\phi_{3c} \ll 1$, $g \approx -G(G + 2)/2$ and it changes its sign at $G = 0$ and $G = -2$. A large negative $h$ enhances the discontinuity of the volume change as observed in Fig. 2 (b). If $\phi_{3c} \ll 1$, $h$ is expressed as $h \approx G^2(G + 3)/3$ and it changes its sign at $G = -3$. Thus, we can possibly induce the VPT even in a gel, which originally shows a continuous volume change, by adding solute of $-G \gg 3$. The gap of the VPT disappears as $\Delta \chi_{31} \propto \sqrt{\phi_{2s} - \phi_{2s}}$, where $\phi_{2s} = (\beta - \phi_{3c}^{(0)})^{-1}$. This disappearing behavior is observed in Fig. 4.

In this paper, only the numerical solutions for $\chi_{12} = 0.0$ are presented. However, we confirmed that the essentially same features are observed for any set of $\chi_{12}$ and $\chi_{31}$ if the resultant $G$ is the same.

### B. The second volume phase transition

In Fig. 7, we plot the swelling curves in a wider range of $\chi_{31}$. We set $\phi_{3c} = 5.0 \times 10^{-2}$, $\nu_{20} = 1.0 \times 10^{-2}$, $B = 0.75$, $\chi_{12} = 0.0$ and $\chi_{32} = -16.0$. It is indicated that another discontinuous volume phase transition can occur at $\chi_{31}$ larger than $\chi_{31t}$ corresponding to the first transition. We confirmed that this second phase transition is observed even if $\nu = 0$ (or, $\alpha = \beta = 0$), while the first one disappears. This fact indicates that this second instability is independent of the network elasticity and has a physical mechanism different from those for well-studied volume phase transitions.

Scott reported that various types of phase diagrams are realized for ternary mixtures (polymer solutions in binary mixtures). They can have several critical points, below which three phases coexist. Analogous to the ternary mixtures, we consider that the second volume transition observed in Fig. 7 is attributed to the bulk instability of the mixing free energy. Since there are some differences between ternary mixtures and gels in mixture solvents, the Scott’s argument cannot be simply applied to the gel systems. The most important differences are that a gel has the elasticity and never exhibits a one-phase homogeneous state.

We have not obtained a simple explanation on the second transition, since non-linearities, which are hard to treat analytically, would play important roles in the bulk instability. In Fig. 7, we also draw the swelling curve obtained by Eq. (21), which neglects the non-linearity of $\phi_{2s}$. The approximated curve also exhibits the second transition, although it does not coincide quantitatively with the numerical solutions for large $\phi_3$. However, this fact implies that the non-linearity of $\phi_2$ has a minor contribution to the essential mechanism of the second transition.

As discussed above, the gel becomes mechanically unstable when $\partial \chi_{31}/\partial \phi_3 < 0$. In Fig. 8, we plot $\partial \chi_{31}/\partial \phi_3$ and $-h(\phi_3, G)\phi_{2s}$, where $\chi_{31}^{(0)}(\phi_3)$ is the swelling curve of the gel in the solvent of the first component. In the range satisfying $\partial \chi_{31}/\partial \phi_3 < -h(\phi_3, G)\phi_{2s}$, the gel would exhibit a discontinuous volume change. It is indicated that strong non-linearity of $h(\phi_3, G)$ with respect to $\phi_3$ is a possible origin of the second transition.
instability is observed even if the elasticity is negligible. Other volume phase transition can occur. This second behavior is well described by the simplified theory in the absence of the additive, by dissolving the strongly good solvent in the gel, which does not exhibit the discontinuous gap in the volume phase transition. Further, we can induce the volume phase transition in the gel. From the simplified theory, the dependence of the transition point on the additive concentration is found to be proportional to \( \partial \chi_{31} / \partial \phi_3 \approx g(\phi_{3c}, G) \), which is approximated as \( g \approx -G(G + 2)/2 \) for a hyper swollen gel.

On the other hand, the difference of the composition becomes large when the additive selectivity is strong. Owing to the non-linearity of the composition difference, the gel tends to be shrunken as the additive concentration is increased, regardless of whether it is good or poor. From the simplified theory, the dependence of the transition point on the additive concentration is found to be proportional to \( \partial \chi_{31} / \partial \phi_3 \approx g(\phi_{3c}, G) \), which is approximated as \( g \approx -G(G + 2)/2 \) for a hyper swollen gel.

(iv) By dissolving the strongly poor additive, we can extinguish the discontinuous volume phase transition. Furthermore, we can induce the volume phase transition in the gel, which does not exhibit the discontinuous gap in the absence of the additive, by dissolving the strongly good additive. In the vicinity of the (first) tri-critical point, this behavior is well described by the simplified theory as shown in Fig. 6 (b).

(iii) Far from the transition point, we found that another volume phase transition can occur. This second instability is observed even if the elasticity is negligible. The transition is caused by the mixing instability. Our theory indicates that the non-linearity of \( \phi_3 \), which is well described by Eq. (27), does have a major contribution to the transition.

Here, we make some remarks about our results.

1) If only van der Waals interaction is taken into account, the interaction parameters should be positive, i.e., \( \lambda_{ij} \geq 0 \). However, a variety of molecular forces, such as ion-dipole interaction, hydrogen bonding and hydrophobic interaction, would also influence the volume phase transition of the gel. When studying their contributions explicitly, we have to consider the microscopic degrees of the freedom in a more specific manner\(^{21,33,34}\). In this work, we assume that the macroscopic degrees of freedom can have a large negative value, by renormalizing these microscopic degrees into them. In a future work, we would like to investigate the connections of such microscopic interactions and our phenomenological theory. Effects of the selective salts\(^{28}\) on polyelectrolyte gels are also interesting.

2) There are a lot of experiments dealing with NIPA gels and PAA gels to study the effects of the additive on the volume phase transition. Most of additives shrink NIPA gels in agreement with our results. Since only a few experiments have reported the difference of the additive concentrations between in and out of the gel\(^{22,35}\), however, we cannot verify the quantitative validity of our model yet. Further experimental studies are highly desired.

IV. SUMMARY AND REMARKS

We studied the volume phase transition of gels immersed in binary mixtures, based on the three-component Flory-Rehner model. Assuming that the volume fraction of the second solvent component is small, we reformulated the Flory-Rehner model into a simple model with a new parameter \( G(\chi_{23} - \chi_{31} - \chi_{12}) \). From numerical solutions and the simplified theory, we found the following behaviors of the volume phase transition of the gel.

(i) When the selectivity of the second component to the polymer network is small, the composition of the mixture solvent in the gel is close to that out of the gel. Here, the single liquid approximation works well. The renormalized interaction parameter between the solvent and the network depends linearly on the composition of the binary mixture. Thus, we can swell the gel by dissolving the additive good to the network, whereas, the gel is shrunken when the poor additive is dissolved.

On the other hand, the difference of the composition becomes large when the additive selectivity is strong. Owing to the non-linearity of the composition difference, the gel tends to be shrunken as the additive concentration is increased, regardless of whether it is good or poor. From the simplified theory, the dependence of the transition point on the additive concentration is found to be proportional to \( \partial \chi_{31} / \partial \phi_3 \approx g(\phi_{3c}, G) \), which is approximated as \( g \approx -G(G + 2)/2 \) for a hyper swollen gel.

(iv) By dissolving the strongly poor additive, we can extinguish the discontinuous volume phase transition. Furthermore, we can induce the volume phase transition in the gel, which does not exhibit the discontinuous gap in the absence of the additive, by dissolving the strongly good additive. In the vicinity of the (first) tri-critical point, this behavior is well described by the simplified theory as shown in Fig. 6 (b).

(iii) Far from the transition point, we found that another volume phase transition can occur. This second instability is observed even if the elasticity is negligible.

The transition is caused by the mixing instability. Our theory indicates that the non-linearity of \( \phi_3 \), which is well described by Eq. (27), does have a major contribution to the transition.

Here, we make some remarks about our results.

1) If only van der Waals interaction is taken into account, the interaction parameters should be positive, i.e., \( \lambda_{ij} \geq 0 \). However, a variety of molecular forces, such as ion-dipole interaction, hydrogen bonding and hydrophobic interaction, would also influence the volume phase transition of the gel. When studying their contributions explicitly, we have to consider the microscopic degrees of the freedom in a more specific manner\(^{21,33,34}\). In this work, we assume that the macroscopic degrees of freedom can have a large negative value, by renormalizing these microscopic degrees into them. In a future work, we would like to investigate the connections of such microscopic interactions and our phenomenological theory. Effects of the selective salts\(^{28}\) on polyelectrolyte gels are also interesting.

2) There are a lot of experiments dealing with NIPA gels and PAA gels to study the effects of the additive on the volume phase transition. Most of additives shrink NIPA gels in agreement with our results. Since only a few experiments have reported the difference of the additive concentrations between in and out of the gel\(^{22,35}\), however, we cannot verify the quantitative validity of our model yet. Further experimental studies are highly desired.

ACKNOWLEDGMENTS

We acknowledge valuable discussions with Akira Onuki. This work was supported by the JSPS Core-to-Core Program “International research network for non-equilibrium dynamics of soft matter” and KAKENHI. The computational work was carried out using the facilities at the Supercomputer Center, Institute for Solid State Physics, University of Tokyo.

1P. J. Flory and J. Rehner, J. Chem. Phys. 11, 512 (1943); ibid 521 (1943).
2K. Daiek and D. Patterson, J. Polym. Sci. A-2. 6, 1209 (1968).
3T. Tanaka, Phys. Rev. Lett. 40, 830 (1978).
4T. Tanaka, D. Fillmore, S.-T. Sun, I. Nishiö, G. Swislow, and A. Shah, Phys. Rev. Lett. 45, 1636 (1980).
5T. Tanaka, Sci. Am. 244, 124 (1981).
6Y. Hirokawa and T. Tanaka, J. Chem. Phys. 81, 6379 (1984).
7I. Oihime and T. Tanaka, J. Chem. Phys. 77, 5725 (1982).
8I. Rëck and T. Tanaka, Macromolecules 17, 2916 (1984).
9A. Fernández-Nieves, A. Fernández-Barbero, and F. J. de las Nieves, J. Chem. Phys. 115, 7644 (2001).
10T. Amiya, Y. Hirokawa, Y. Hirose, Y. Li, and T. Tanaka, J. Chem. Phys. 86, 2375 (1987).
11S. Hirotsu, Y. Hirokawa, and T. Tanaka, J. Chem. Phys. 87, 1392 (1987).
12C. G. Park and A. S. Hoffman, Macromolecules 26, 5045 (1993).
13H. Inoue, S. Goto, K. Otake, and C. Saito, Langmuir 8, 687 (1992).
14 M. Annaka, K. Motokawa, S. Sasaki, T. Nakahira, H. Kawasaki, H. Maeda, Y. Amo, and Y. Tominaga, J. Chem. Phys. 113, 5980 (2000).
15 S. Hirotsu, J. Phys. Soc. Jpn. 56, 233 (1987).
16 E. Kokufuta, Y.-Q. Zhang, T. Tanaka, and A. Mamada, Macromolecules 26, 1053 (1993).
17 H. Kawasaki, S. Sasaki, H. Maeda, S. Mihara, M. Tokita, and T. Komai, J. Phys. Chem. 100, 16272 (1996).
18 S. Sasaki, H. Kawasaki, and H. Maeda, Macromolecules 30, 1847 (1997).
19 D. Dhara and P. R. Chatterji, Langmuir 15, 930 (1999).
20 S. Koga, S. Sasaki, and H. Maeda, J. Phys. Chem. B 105, 4105 (2001).
21 F. Tanaka, T. Koga, and F. M. Winnik, Phys. Rev. Lett. 101, 028302 (2008).
22 T. Ishidao, Y. Hishimoto, Y. Iwai, and Y. Araki, Colloid Polym. Sci. 272, 1313 (1994).
23 V. V. Vasilevskaya, V. A. Ryabina, S. G. Starodubtsev, and A. R. Khokhlov, Polym. Sci. U.S.S.R. 31, 784 (1989).
24 T. Iwatsubo, K. Ogasawara, A. Yamasaki, T. Masuoka, and K. Mizoguchi, Macromolecules 28, 6579 (1995).
25 W. R. Krigbaum and D. K. Carpenter, J. Polym. Sci. 14, 241 (1954).
26 G. M. Bristow, Trans. Faraday Soc. 55, 1246 (1959).
27 O. Okeowa and J. R. Dorgan, Macromolecules 39, 8193 (2006).
28 R. Okamoto and A. Onuki, Phys. Rev. E 82, 051501 (2010).
29 A. Onuki and R. Okamoto, Curr. Opin. Colloid Interface Sci. 16, 525 (2011).
30 A. Onuki, R. Okamoto, and T. Araki, Bull. Chem. Soc. Jpn. 23, 284113 (2011).
31 R. L. Scott, J. Chem. Phys. 17, 268 (1949).
32 P. J. Flory, Principle of Polymer Chemistry (Cornell Univ. Press, Ithaca, New York, 1953).
33 H. Kojima and F. Tanaka, Macromolecules 43, 5103 (2010).
34 H. Kojima and F. Tanaka, Soft Matter 8, 3010 (2012).
35 T. Iwatsubo and T. Shinbo, J. Macromol. Sci.-Physics. B 40, 1017 (2001).