Formation of pale foam glass from colored glass cullet

Tomoko AKAI¹, Kohei FUKUMI¹,† and Masaru YAMASHITA¹

¹Inorganic Functional Material Research Institute, National Institute of Advanced Industrial Science and Technology, 1–8–31 Midorigaoka, Ikeda, Osaka 563–8577, Japan

The decolorization of foam glasses made from green and amber glass cullets was attempted. It was found that the addition of silicon carbide (SiC) and zinc oxide (ZnO) as decolorizing agents was effective for the decolorization of both the foam glasses made from green and amber glasses. Therefore, the addition of SiC and ZnO was suitable for the decolorization of foam glass made from the mixture of green, amber and colorless glasses.

Key-words: Foam glass, Glass recycle, Decolorization, Zinc oxide, Silicon carbide

1. Introduction

Used glass bottles are recycled into products such as bottles, short glass fibers, and aggregates for subgrade and base. More than 90% of colorless and amber (brown) glass bottles are recycled as raw materials of glass bottles. Only 40% of glass containers in other colors are, however, recycled into glass bottles in Japan. Consequently, the cullet price of green and mixed glasses (the mixture of amber, green and colorless glasses) is lower than that of colorless and amber glass. Therefore, the cullet of green and mixed glasses is a cost-effective material and is suitable for use of civil engineering materials.

Foam glass is one of the usage of the mixed cullet. Many studies have been carried out on the foaming process, the reaction mechanism between glass and foaming agent and the effect of preparation conditions, such as foaming temperature, foaming time, particle size and type of foaming agent, on the properties of foam glasses. Foam glasses are used as a lightweight aggregate, an aggregate for heat insulation blocks, a drain-improving material for agriculture, a crime-preventive gravel, a crime-preventive gravel. White crime-preventive gravel is preferable using white foam glass on a ground face, such as cable only to opaque glasses such as foam glass. In the present study, we have attempted to produce white foam glass from the cullets of amber glass, green glass and the mixture of colorless, amber and green glasses.

It is known that the coloration of amber glass is caused by the charge transfer electronic transition in $[\text{Fe}^{3+}\text{O}^{2-}\cdot\text{S}^{2-}]^{3-}$ chromophore. Decolorization of amber glass has been realized by adding zinc oxide (ZnO) to amber glass cullet during the remelting of glass, since sulfide ions and oxide ions are exchanged between $[\text{Fe}^{3+}\text{O}^{2-}\cdot\text{S}^{2-}]^{3-}$ chromophore and ZnO during remelting and the resultant Fe$^{3+}$ ions coordinated with only oxide ions and ZnS exhibit weak or no absorption in the visible ray region. Foam glass is, however, produced at lower temperatures than the melting temperature (liquidus temperature), for the retention of bubbles in glass. Therefore, it is important to study the decolorization of amber glass at temperatures lower than the melting temperature at first. The coloration of green glass is caused by the d–d transition in Cr$^{3+}$ ions. The decolorization of green glass has not been studied so far, since divalent, trivalent, tetravalent and hexavalent Cr ions, all exhibit optical absorption in visible ray region. Therefore, it is also important to develop the decolorization method of green glass. One possible method of decolorization is the incorporation of light scatterers such as fine bubbles in glasses, which would lighten colors, although this method is applicable only to opaque glasses such as foam glass.

In the present study, we have attempted to produce white foam glass from the cullets of amber glass, green glass and the mixture of colorless, amber and green glasses.

2. Experimental procedures

Waste glass cullets of colorless, amber and green glasses were obtained from waste traders. The viscosity of the colorless, amber and green glasses was measured with an indentation, parallel-plate, and parallel-plate-rotation-type viscometer (PRVM-1100, OPT Corp.). The equal weight mixture of the colorless, amber and green glasses was prepared as a “mixed glass”. The glass cullets of amber, green and mixed glasses were crushed to a particle size less than 100 μm. Three grams of crushed cullet was mixed with 0.5 mass % of CaCO₃ powder (reagent grade, Wako) as a foaming agent, and 0 to 2 mass % of ZnO (reagent grade, Kishida) and 0 to 0.4 mass % silicon carbide (SiC) (#8000, Green Carbo, Fujimi) as decoloring agents. The mixture
was put in an alumina crucible, heated up to 850 °C at a heating rate of 9.2 °C/min, kept at 850 °C for 30 min and allowed to cool, in an ambient atmosphere within an electric furnace, to obtain foam glasses. The foam glasses so obtained were observed with a digital microscope (VHX-500, Keyence). The L*a*b* color coordinates were measured at bottom surfaces of foam glasses with a colorimeter (CM-600d spectrophotometer, Konica Minolta), and were converted to L* C*h* coordinates, since the higher brightness (L*) and lower metric chroma (C*) are desirable color change. The measurement errors of L*, C* and h* were within 0.7, 0.3 and 0.5, respectively. X-ray diffraction measurement was carried out for the foam glasses made from the green glass. The density and porosity of a foam glass made from green glass cullet was measured by the Archimedes method using water as the immersion liquid. Electron-spin resonance (ESR) spectra of glass samples without a foaming agent were measured with a conventional X-band ESR spectrometer (300E ESR spectrometer, Bruker). The appropriate mixture of SiO2 (Cerac, 99.99%), Na2CO3 (Asahi Glass, 99.97%), CaCO3 (Cerac, 99.999%) and Al2O3 (Cerac, 99.99%) was melted in a Pt crucible to form 16Na2O–10CaO–1Al2O3–73SiO2 glass. The glass was crushed into powder, mixed with 2 mass % ZnO and 0.2 mass % SiC, and heated at 850 °C for 30 min. The glass sample so obtained was cut to dimensions of 2 × 2 × 25 mm for ESR measurement. The measurement was carried out at a 5-mW power, 0.5-mT modulation amplitude, and 100-kHz frequency, at room temperature. The g-value was corrected by measuring a strong pitch.

3. Results and discussion

Figure 1 shows the viscosity-temperature curves of colorless, amber and green glasses. These curves overlapped each other, showing that these glasses can be heat-treated at the same temperature for the production of foam glass. Since it is known that the viscosity range for the formation of foam glass is 106 to 104 dPa s4) and that CaCO3 is decomposed to evolve gaseous CO2 at temperatures ranging from 600 to 850 °C,20 the heat treatment was carried out at 850 °C in the present study. All the foam glasses had bubbles of diameters less than 1 mm, as shown in Figs. 2 and 3. The true density, apparent density, bulk density, total porosity, open porosity and closed porosity were 2.55 ± 0.01, 0.59 ± 0.01 and 0.51 ± 0.01 g cm−3, 79 ± 1, 13 ± 1 and 66 ± 1%, respectively, in the foam glass made from green glass cullet with 2 and 0.2 mass % SiC.

3.1 Green glass cullet

L* C*h* values of foam glasses made from green glass cullet are depicted in Fig. 4. The sole addition of ZnO as a decoloring agent slightly increased L* (brightness), although C* (chroma) and h (hue angle) were scarcely changed. Since ZnO has refractive index higher than the glass and ZnO is transparent in the visible ray region, it is deduced that ZnO acts as simple light scatterers in the foam glass, which increased L*. On the other hand, unexpectedly, the sole addition of SiC caused the change in h from 128 to 90°, the decrease in L* and the increase in C*, indicating that the color changed from green to yellow and became deeper by the addition of SiC. The addition of both SiC and ZnO caused the change in h from 128 to 111°, the increase in L* and the decrease in C*, indicating that the color changed from green to lime green and became paler. The foam glass in which 2 mass % ZnO and 0.1 mass % SiC were contained looks pale lime green color, although the foam glass with no additives looks green color, as shown in Fig. 2.

Figure 5 shows the X-ray diffraction patterns of the green foam glasses. It can be seen that partial crystallization occurred and that the diffraction patterns were almost independent of the addition of SiC and ZnO, showing that the color change by the addition of SiC was not due to the partial crystallization. Figure 3 shows the optical microscopic images of the glasses. Much more fine bubbles with a size less than about 10 μm were formed in the glasses in which SiC was added than in the glasses in which SiC was not added. Therefore, it is deduced that the addition of SiC causes many fine bubbles through the oxidation reaction, which acts as light scatterers. Although SiC is often used as a foaming agent which forms large bubbles in glasses.
after heating at temperatures higher than 1000 °C, fine bubbles were formed by the addition of SiC owing to the lower heating temperature in the present study. It is unlikely, however, that fine bubbles cause the color change. It is expected, rather, that fine bubbles increase only $L^*$, similar to ZnO, in contrast to the experimental results. This discrepancy was discussed in the following section.

Extensive studies have been performed on the oxidation of SiC and the reaction between SiC and glasses. It has been shown that C, CO or CO$_2$ is formed as well as SiO and SiO$_2$ at the interface between SiC and glasses. According to the result of ESR measurement shown in Fig. 6, an ESR signal was observed at $g \sim 2.0025$ in the glass in which SiC was added, although such a signal was not observed in the glass in which SiC was not added. Since the $g$ value of this signal was close to that of dangling bonds of carbon atoms in amorphous carbon observed in SiO$_2$/SiC interface, the presence a signal at $g \sim 2.0025$ shows the formation of carbon layer in the present glass. From the view point of absorption coef-
ficient (the absorption coefficients of amorphous carbon, green SIC and SiO were 1.9 × 10^5, 96 and 2.2 × 10^3 cm^{-1} at a wavelength of 450 nm and 1.1 × 10^3, 94, and 1.5 × 10^2 cm^{-1}, at a wavelength of 650 nm, respectively), carbon has the highest coefficient. Since the densities of glass and amorphous carbon are 2.55 and 2.1 g cm^{-3}, respectively, it is expected that the transmittance for 0.1 mm thickness at 650 nm is 5 percent higher than that at 450 nm, assuming that only 10% of SIC is converted to amorphous carbon. Therefore, carbon probably acted as light absorbers, which changed color and decreased L^* in the foam glass.

The addition of ZnO suppressed the effect of SIC on the values of L^* and h in the foam glass in which both SIC and ZnO were added. This suppression is due to the decrease of the fraction of light which arrives at carbon. On the other hand, it is unclear why the co-addition of SIC and ZnO reduced C^*.

### 3.2 Amber glass cullet

L^*C^*h values of foam glasses made from amber glass cullet are depicted in Fig. 4. The sole addition of ZnO increased L^* and decreased C^*, indicating the decolorization of the glass. In addition, h changed from 83 to 100°, that is, the color changed from amber to green. These findings indicate that the reaction between [Fe^{3+}O^{2-};S^{2-}]^{5-} chromophore and ZnO occurred at a temperature of 850 °C to form [Fe^{3+}O^{2-};S^{2-}]^{3-} and ZnS. The green color was due to the Fe^{2+} ions which coexisted with Fe^{3+} ions in the glass. Since it is known that the chromophore starts to decompose at a temperature of 550 °C and most sulfur exists as free S^{2-} at temperatures higher than 700 °C, it is expected that the reaction rate is controlled by the diffusion of sulfur in the glass. Since it has been reported that the diffusion coefficient of sulfur in glasses primarily depends on the viscosity of glasses, the diffusion coefficient D at η = 10^{-5} Pas (850 °C) was estimated to be 3 × 10^{-14} m^2 s^{-1}. The approximate diffusion length of sulfur [(Dτ)^{1/2}] was 7.5 μm, which was one order magnitude smaller than the maximum size of the glass particles. The formation of bubbles, however, caused the thinning of glass layers surrounding bubbles, which reduced the diffusion path length required for sulfur. Therefore, it was important for the reaction between the chromophores and ZnO to form bubbles in the present study. In fact, the elimination of CaCO_3 reduced the value of h from 100 to 91°. It is known that the content of sulfur in ambers glasses is about 0.025 mass%, which corresponds to 8 × 10^{-6} mol S/1 g glass. Since the content of sulfur in amber glasses was much lower than the addition amount of ZnO (2 × 10^{-3} mol ZnO/1 g glass), it is deduced that the unreacted ZnO acts as light scatterers.

In the green glass, it was found that the green color was reduced by the addition of SiC. Further addition of SiC to the ZnO added glass reduced C^*, as expected from the result of green glass, that is, the green color was reduced. Although the addition of SiC might have reduced Fe^{3+} to Fe^{2+} and increased green color, such an effect was not observed. The foam glass in which 2 mass% ZnO and 0.1 mass% SiC were contained looks pale lime green color, although the foam glass with no additives looks amber, as shown in Fig. 2.

### 3.3 Mixed glass cullet

As mentioned above, foam glasses made from amber glass cullet and green glass cullet were, both, decolorized by the addition of the same additives as decolorizing agents. L^*C^*h values of foam glasses made from mixed glass cullet are depicted in Fig. 4. As expected from the results of green and amber glass cullets, the foam glass made from mixed cullet was decolored by the addition of ZnO and SiC. Since the mixed glass included the colorless glass, the value of L^* in the foam glass made from the mixed cullet was higher than those made from green and amber glass cullets.

### 4. Summary

We have succeeded in decolorizing foam glasses made from amber glass cullet, green glass cullet and mixed glass cullet. ZnO and SIC was effective as decolorizing agents for the foam glasses made from green glass cullet, amber glass cullet and mixed glass cullet. It is deduced that ZnO acts as mainly white scatterers and decomposers of chromophores in the foam glasses made from green glass cullet and amber glass cullet, respectively. SiC acted as absorbers of light in these foam glasses.

**Acknowledgment** The authors are grateful to M. Murakami and S. Matsumoto for their assistance in the experiments.

**References**

1) Glass Bottle 3R Promotion Association, Data collection, http://www.glass-3r.jp/data/pdf/data_10.pdf, accessed 31/01/2018 [in Japanese].

2) R. R. Petersen, J. J. König and Y. Yue, *J. Non-Cryst. Solids*, 425, 74–82 (2015).

3) J. J. König, R. R. Petersen and Y. Yue, *J. Non-Cryst. Solids*, 447, 190–197 (2016).

4) G. Scarinci, G. Brusatin and E. Bernado, “Cellular Ceramics: Structure, Manufacturing, Properties and Applications”, Ed. by M. Scheffler and P. Colombo, Wiley-VCH, Weinheim (2005) Chapter 2.7, pp. 158–
5) G. Brusatin, E. Bernardo and G. Scarinci, “Sustainable Waste Management and Recycling: Glass Waste”, Ed. by M. C. Limbachiya and J. J. Roberts, Thomas Telford, London (2004) pp. 67–82.

6) J. García-Ten, A. Saburit, M. J. Orts, E. Bernardo and P. Colombo, *Eur. J. Glass Sci. Technol. A*, 52, 103–110 (2011).

7) M. B. Østergaard, R. R. Petersen, J. J. König, H. Johra and Y. Yue, *J. Non-Cryst. Solids*, 465, 59–64 (2017).

8) G. W. McLellan and E. B. Shand, “Glass Engineering Handbook”, 3rd edition, McGraw-Hill, New York (1984) pp. 19–19-5.

9) J. König, R. R. Petersen and Y. Yue, *Ceram. Int.*, 41, 9793–9800 (2015).

10) D. Kazmi, D. J. Williams and M. Serati, *Int. J. Appl. Ceram. Technol.*, 00, 1–26 (2019).

11) S. Fotiadou, M. C. Limbachiya, A. N. Fried and J. J. Roberts, “Sustainable Waste Management and Recycling: Glass Waste”, Ed. by M. L. Limbachiya and J. J. Roberts, Thomas Telford, London (2004) pp. 305–312.

12) L. Hu, F. Bu, F. Guo and Z. Zhang, *Earth Enviro. Sci.*, 61, 012122 (2017).

13) Y. L. Wei, C. Y. Lin, K.-W. Ko and H. P. Wang, *Mar. Pollut. Bull.*, 63, 135–140 (2011).

14) Glass Foam Business Cooperative Brochure of Organization Guide, http://www.supersol.jp/, da0c6e71b225968600f6c75dc0ea8a35.pdf, accessed 17/12/2019 [in Japanese].

15) R. W. Douglas and M. S. Zaman, *Phys. Chem. Glasses*, 10, 125–132 (1969).

16) M. M. Morsi, S. I. El-sherbiny and K. M. Mohamed, *Spectrochim. Acta A*, 145, 376–383 (2015).

17) D. B. Rapp, M. A. Dorsey, M. M. Ashton-Patton and J. E. Shelby, *Ceram. Trans.*, 141, 323–330 (2004).

18) T. Murata, M. Torisaka, H. Takebe and K. Morinaga, *J. Non-Cryst. Solids*, 220, 139–146 (1997).

19) A. Paul, *Phys. Chem. Glasses*, 15, 91–94 (1974).

20) C. Rodriguez-Navarro, E. Ruiz-Agudo, A. Luque, A. B. Rodriguez-Navarro and M. Ortega-Huertas, *Am. Mineral.*, 94, 578–593 (2009).

21) R. F. Cooper and K. Chyung, *J. Mater. Sci.*, 22, 3148–3160 (1987).

22) D. Das, J. Farjas and P. Roure, *J. Am. Ceram. Soc.*, 87, 1301–1305 (2004).

23) H. Katsui, M. Oguma and T. Goto, *J. Am. Ceram. Soc.*, 97, 1633–1637 (2014).

24) J. Roy, S. Chandra, S. Das and S. Maitra, *Rev. Adv. Mater. Sci.*, 38, 29–39 (2014).

25) V. V. Afanas’ev, “Defects in SiO2 and Related Dielectrics: Science and Technology”, Ed. by G. Pacchioni, L. Skuja and D. L. Griscom, Kluwer, Dordrecht (2000) pp. 581–597.

26) H. J. von Bardeleben, J. L. Cantin, M. Mynbaeva, S. E. Saddow, Y. Shishkin, R. P. Devaty and W. J. Choyke, “Silicon Nitride and Silicon Dioxide Thin Insulating Films VII”, Proceedings Volume 2003-02, Ed. by R. E. Sah, K. B. Sundaram, M. J. Deen, W. D. Brown, D. Landheer and D. Misra, Electrochem. Soc., Pennington (2003) pp. 39–51.

27) H. Werheit and K. A. Schetz, *J. Solid State Chem.*, 177, 580–585 (2004).

28) S. Logothetidis, *Diamond Relat. Mater.*, 12, 141–150 (2003).

29) H. R. Phillip, *J. Phys. Chem. Solids*, 32, 1935–1945 (1971).

30) C. R. Bamford, “Colour Generation and Control in Glass”, Glass Science and Technology 2, Elsevier, Amsterdam (1977) pp. 35–38.

31) M. Müller, C. Russel and O. Claußen, *Glastech. Ber.*, 72, 362–366 (1999).

32) H. Behrens and J. Stelling, *Rev. Mineral. Geochem.*, 73, 79–111 (2011).

33) Japan Glass Bottle Association http://glassbottle.org/ecology/apply/, accessed 03/12/2019 [in Japanese].

34) R. Falcone, S. Ceola, A. Daneo and S. Maurina, *Rev. Mineral. Geochem.*, 73, 113–141 (2011).

35) ICDD PDF-2 card No. 00-023-671.

36) ICDD PDF-2 card No. 00-039-1425.