Abstract—AlGaN alloys are promising for field emission devices due to their low electron affinities. However, there have been limited demonstrations of AlGaN vacuum transistors so far. This letter combines a new self-aligned-gate (SAG) process and digital-etching tip sharpening to demonstrate three-terminal AlGaN SAG field emitter arrays (FEAs). These devices show a turn-on voltage of 19.5 V and an anode current density ($J_A$) of 100 mA/cm² at an overdrive voltage of 20 V, which are comparable with best Si devices. The AlGaN SAGFEAs can operate in DC mode at a fixed gate-emitter voltage ($V_{GE}$) with $J_A$ of 3-5 mA/cm² for at least 5 hours without a significant degradation. The gate leakage does not increase after the long DC operation, suggesting high-performance and stable AlGaN vacuum transistors.

Index Terms—AlGaN, field emission, vacuum transistor.

I. INTRODUCTION

Vacuum devices are excellent candidates for harsh-environment and high-frequency electronics thanks to their radiation hardness and scattering-free electron transport [1]. Si and metal field emitters (FEs) have been developed into mature technologies in recent years [2]–[5]. III-Nitride semiconductors promise to further improve FEs thanks to their tunable electron affinities [6], [7]. With a lower electron affinity, electrons could tunnel from the semiconductor surface to vacuum more easily, leading to devices with a lower operating voltage. Ga-polar heavily-doped GaN self-aligned-gate field emitter arrays (SAGFEAs) have been demonstrated and improved with sharpened tips by wet-based digital etching (DE) recently [8], [9]. The gate-emitter turn-on voltage ($V_{GE, on}$) in these devices is comparable to the state-of-the-art Si devices, while the relatively low current density and device stability are still unsolved issues. Reference [8] observed that the gate insulator under the probing region breaks when $V_{GE}$ approaches 50 V, while no damage was observed in the FEA region. Layout optimization is thus expected to improve device stability.

In addition to GaN, AlGaN alloys with high-Al ratio are expected to provide better performance than GaN because of reduced electron affinities [6]. However, most work on these materials has focused on two-terminal field emitters [10]–[13]. Very limited demonstrations of AlGaN three-terminal FEAs have been reported [14], [15], even though these devices are necessary for accurate benchmarking with other materials’ systems. Schottky-diode-type electron emission devices based on the low electron affinities of AlGaN alloys have also been demonstrated; however, their power consumption is not competitive with the SAGFEAs [16], [17]. In this work, we have fabricated AlGaN SAGFEAs with a similar SAG geometry and DE recently demonstrated for GaN SAGFEAs [8], [9]. We show a clear performance improvement, compared to simultaneously-fabricated GaN devices. For these AlGaN SAGFEAs, a turn on voltages ($V_{GE, on}$) of 19-22 V and anode current densities ($J_A$) approaching 100 mA/cm² at an overdrive voltage ($V_{OV} = V_{GE} - V_{GE, on}$) of 20 V were demonstrated, which are similar to the best Si devices. Additionally, these AlGaN SAGFEAs show over 5 hours of stable DC operation for anode current density ($J_A$) of 3-5 mA/cm² with no increase of gate leakage. These results show the great potential for III-Nitride devices to outperform Si devices in the future.

II. DEVICE FABRICATION

Baseline GaN SAGFEAs are fabricated on a GaN-on-Si wafer grown by Enkris Semiconductor, Inc. by metal organic chemical vapor deposition (MOCVD) [8]. The structure consists of a 1.4 $\mu$m n$^{++}$-GaN (Si: 1 × 10$^{19}$ cm$^{-3}$) layer on a 1.4 $\mu$m buffer layer grown on the Si substrate. AlGaN SAGFEAs are fabricated on a 1 cm×1 cm coupon grown at the Georgia Institute of Technology. This AlGaN material consists of a 420-nm n$^+$ AlGaN layer and a 200-nm non-intentionally-doped AlN layer, which are grown by molecular beam epitaxy (MBE), on a commercial AlN template [18]–[20]. The AlN template is grown on the sapphire substrate via hydride vapor phase epitaxy (HVPE). The Al composition in this AlGaN
The Hall electron concentration of the n-type material is 37.6%, as measured by x-ray diffraction (XRD). The Hall electron concentration of the n\textsuperscript{+} AlGaIn layer is about 3.9 × 10\textsuperscript{19} cm\textsuperscript{-3}.

Fig. 1(a) shows the fabrication process flow, and the details of different processing steps can be found in prior work [8]. The III-N tips are firstly formed by dry etching and are then sharpened by DE to obtain sub-20 nm tip width [9]. Contrary to GaN, it is observed that the AlGaIn sidewalls become rough after DE (Fig. 2(a)). The rough sidewalls can potentially provide additional electron emission sites and increase gate leakage. After tip formation, an additional 400-nm plasma enhanced chemical vapor deposition (PECVD) SiO\textsubscript{2} is added with respect to our prior work in [8] to prevent early breakdown happening under the gate pad. A 10-nm Al\textsubscript{2}O\textsubscript{3} is then deposited by atomic layer deposition (ALD) to protect III-N tips from subsequent plasma dry etching. In the last step, tips are exposed by a two-step etching. The tetraethyl orthosilicate (TEOS) is firstly etched by timed dry etching, and the Al\textsubscript{2}O\textsubscript{3} and residual TEOS are then etched by short-dipping the sample in buffer oxide etchant (BOE). The scanning electron microscopy (SEM) images of a finished AlGaIn FEA with sub-20-nm tip width are shown in Fig. 2(b) and (c).

### III. Results and Discussion

After SEM inspection, samples are loaded into an ultra-high vacuum (UHV) system for measurement at a base pressure of 1-2 × 10\textsuperscript{-9} Torr. The anode is a suspended 0.5-mm-diameter tungsten ball which can be moved to position on top of the measured FEA. The anode-emitter distance (d\textsubscript{AE}) is kept around 2 mm in all measurements in this work.

The transfer and output characteristics after proper conditioning of one of the best AlGaIn SAGEFEAs are shown in Fig. 3. This AlGaIn FEA consists of 150 × 150 tips, and the FEA area is about 96 µm × 80 µm. The transfer characteristics are stable as there is no clear deviation between 3 sequential measurements. The device turns on at a gate-emitter voltage (V\textsubscript{GE}) of 19.5 V at which point the anode current (I\textsubscript{A}) increases to 10 pA from the noise level. The maximum anode current is 8 µA, which corresponds to a current density J\textsubscript{A} of about 100 mA/cm\textsuperscript{2}. Moreover, though the AlGaIn sidewalls are rough after DE (Fig. 2(a)), the gate leakage (I\textsubscript{G}) is still at least one-order-of-magnitude lower than the anode current. The Fowler-Nordheim (F-N) plot of I\textsubscript{A} shows a clear straight line with a slope (−b\textsubscript{FN}) of −439.7 V/µm (Fig. 3(b)). The output characteristics of this AlGaIn device shows a stable anode current for anode-emitter voltages (V\textsubscript{AE}) above 100 V. This large V\textsubscript{AE} is due to space-charge limit (Child’s law) [21], [22], and it can be reduced by integrating the anode into the device structure and reducing, in that way, the anode-emitter distance, d\textsubscript{AE} [23]. The device is stable and I\textsubscript{G} does not increase after the measurements.

A different AlGaIn SAGEFA with 150 × 150 tips is measured for both transfer characteristics and DC operation lifetime (Fig. 4). Based on the transfer characteristics (Fig. 4(a)), the device does not show significant degradation after a total 5-hr DC operation at a fixed V\textsubscript{GE} of 37 V (Fig. 4(b)). At the beginning of the lifetime test, I\textsubscript{A} was about 400 nA, which corresponds to a current density J\textsubscript{A} of about 5 mA/cm\textsuperscript{2}. Both I\textsubscript{A} and I\textsubscript{G} gradually decrease during the lifetime test, and the measurement stops when the I\textsubscript{A} goes below 200 nA. F-N parameters for multiple transfer characteristics before and after lifetime test are plotted in the Seppen-Katamuki (S-K) chart (Fig. 4(c)). As the data points stay in the same region of the S-K chart, it is clear that the device does not degrade significantly [24]. The variability during the lifetime measurement could be related to noise introduced by micro-vibrations happening during the measurement, as the anode is a suspended metal ball in the measurement system, which is not fixed to the device.

The F-N parameters of 4 different AlGaIn SAGEFEAs and a baseline GaN device fabricated through the same process are summarized and compared with our prior work in the S-K chart shown in Fig. 5(a). Low |b\textsubscript{FN}| and large ln(a\textsubscript{FN}) are desired for lower operating voltage and higher anode current. Although the performance of the baseline GaN device is slightly worse than prior work [8], the changes in device...
fabrication allow the new GaN device to be more stable than the one in reference [8], and the early breakdown in the insulator layer under the probing region is eliminated. Furthermore, the AlGaN devices clearly show better performance than GaN devices thanks to their potentially-lower work function. The effective work function is typically lower if the data points are located at a more upper-right region in the S-K chart [24]. Ideally, heavily doped n-type AlGaN alloy with a higher Al composition ratio should provide better performance because of a lower electron affinity. However, surface states, doping activation energies, and defects, which are generated during growth and device fabrication, are related to the Al composition ratio and will affect the surface effective work function of the fabricated tips [9], [25]. Moreover, high-Al-composition AlGaN materials are prone to form oxide in air, which will also change the energy barriers for electron field emission. Therefore, there might be an optimum Al composition ratio of the AlGaN alloy for the field emission devices, while it still requires more detailed study on different aspects mentioned above.

The III-N SAGFEAs demonstrated in this work are also compared with SAGFEAs based on other materials (Fig. 5(b)) [3], [4], [8], [26]–[28]. When setting an overdrive voltage (\( V_{VQ} = V_{GE} - V_{GE,ON} \)) of 20 V, the \( J_A \) of our AlGaN SAGFEAs approaches 100 mA/cm² and is comparable with state-of-the-art Si SAGFEAs. Based on the F-N parameter (\( b_{FN} = 439.7 \) V), our AlGaN SAGFEAs can potentially outperform Si devices. However, the defects introduced in the AlGaN devices by the dry etching and DE (Fig. 2(a) and (b)) make these devices prone to break at higher \( V_{GE} \). A strong electric field at high \( V_{GE} \) in the oxide layer covering the unexposed tips (as shown in the cross-section geometry in Fig. 1) can also cause gate oxide breakdown. These unexposed tips are necessary to extend out the gate metal from the FEA region to the gate pad region, and further optimization is necessary to mitigate or eliminate this failure mechanism. With further optimization on different etching process and device geometry, the device is expected to be more stable and to be able to support a higher gate voltage. More detailed study on device stability and lifetime at different current densities can then be done for better III-N SAGFEAs in the future.

### IV. CONCLUSION

Three-terminal 37.6%-Al AlGaN SAGFEAs are demonstrated with a turn-on voltage of 19.5 V at \( I_A = 10 \) pA and a \( J_A \) of 100 mA/cm² at \( V_{GE} = 40 \) V. Thanks to the lower effective work function, the operation voltage and current density of AlGaN SAGFEAs are better than GaN devices, and competitive with state-of-the-art Si devices. These AlGaN SAGFEAs can operate in DC mode with \( J_A \) of 3-5 mA/cm² at a fixed \( V_{GE} \) for 5 hours without a clear degradation, suggesting stable field emission devices. Though further study on material properties and fabrication optimization is needed, these results show that AlGaN SAGFEAs have great potential for high-frequency and harsh-environment applications.

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The GaN-on-Si coupon used in this work was obtained from the same 6-inch wafer reported in prior work [8], provided by Enkris, Inc. The AlGaN materials were grown by MBE on AlN templates and characterized at the Georgia Institute of Technology. The device fabrication and characterization was conducted in MTL, NSL, and MIT.nano at MIT.

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