Formation of nanostructured silicon surfaces by stain etching

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Abstract

In this work, we report the fabrication of ordered silicon structures by chemical etching of silicon in vanadium oxide (V2O5)/hydrofluoric acid (HF) solution. The effects of the different etching parameters including the solution concentration, temperature, and the presence of metal catalyst film deposition (Pd) on the morphologies and reflective properties of the etched Si surfaces were studied. Scanning electron microscopy (SEM) was carried out to explore the morphologies of the etched surfaces with and without the presence of catalyst. In this case, the attack on the surfaces with a palladium deposit begins by creating uniform circular pores on silicon in which we distinguish the formation of pyramidal structures of silicon. Fourier transform infrared spectroscopy (FTIR) demonstrates that the surfaces are H-terminated. A UV-Vis-NIR spectrophotometer was used to study the reflectance of the structures obtained. A reflectance of 2.21% from the etched Si surfaces in the wavelength range of 400 to 1,000 nm was obtained after 120 min of etching while it is of 4.33% from the Pd/Si surfaces etched for 15 min.

Keywords: Silicon; Etching time; Vanadium oxide; Catalyst

Background

In the current semiconductor industry, nanostructures of silicon represent the basic material for the conception of several devices in the field of nanoelectronics [1,2], optoelectronics [3], energy conversion [4,5], energy storage [6,7], and also (bio)chemical sensors [8,9]. Various methods have been developed to fabricate Si nanostructures such as reactive-ion etching (RIE), electrochemical etching, metal-assisted etching, or stain etching. This last one is an electroless method of forming porous silicon (PSi) in a mixture based on hydrofluoric acid (HF) and an oxidant. The nanostructuring of silicon by stain etching has attracted increasing attention in recent years for several reasons. One of these reasons is that it is an inexpensive method with the ability to control various parameters and can be accomplished in a simple chemical laboratory. The most widely used oxidant is nitric acid (HNO3) which has been investigated in numerous studies for the development of silicon nanostructures (nanopillars and nanowires) for their interesting fields of application [10]. However, the use of HNO3 leads to bubble formation, inhomogeneous films, and irreproducible results.

Recently, Kolasinski and Barclay demonstrated that using vanadium oxide (V2O5) in the etching solution seems to be a good way to avoid bubble formation (no gas is generated). V2O5 dissolves in HF(aq) to form VO2+ which according to the half reaction \( \text{VO}^+ + \text{H}^+ + e^- \rightarrow \text{VO}^{2+} + \text{OH}^- \) with an appropriate acceptor level at \( E_0 = +1.0 \text{ V} \), which is able to inject holes into the Si valence band [11]. Several studies on the kinetics of the stain etching using V2O5 have been demonstrated, but few works on the morphological properties of the structures have been obtained.

In this work, we report the fabrication of ordered array pillar silicon and silicon macropores by a simple chemical etching of silicon in vanadium (V) oxide/hydrofluoric acid solution. Different etching parameters including the solution concentration, temperature of Si substrates, and thin metal catalyst film deposition (Pd) on the Si surface were studied. The etched surfaces were characterized by scanning electron microscopy and spectrophotometry.
Methods

Twenty-nanometer palladium (Pd) films are deposited by evaporation technique, on one side of p-type (1 to 10 Ω cm) single crystal (1 0 0) silicon wafers (both sides polished), with 250 to 300 μm thickness, cleaned using a dilute HF aqueous solution (1:10) for 30 s prior to deposition. Deposition of a significantly thinner Pd film can result in uniform etching of the entire silicon surface during chemical etching. The last step of the fabrication process consists of etching the Si and the Pd/Si samples in a mixture of HF (49%; Sigma-Aldrich, St. Louis, MO, USA) and V₂O₅ (98%; Sigma-Aldrich), for a period of 30, 90, and 120 min. Only one side is in contact with the etchant by using an adapted cell, and the edges are protected by an O ring.

After etching, the samples are thoroughly rinsed in DI water and dried with nitrogen stream. The etch rate was obtained by dividing the etch amount, i.e., Si weight before and after the etching process of Si by reaction time, exposed surface area, and Si density (2.33 × 10³ kg/m³). Photoluminescence analyses were performed with a PerkinElmer LS 50B (PerkinElmer, Waltham, MA, USA) spectrometer.

The UV-Visible reflectance of the etched silicon was measured using a Cary 500 Varian spectrophotometer (Varian Medical Systems Inc., Palo Alto, CA, USA) in the range of 400 to 1,100 nm. The surface morphology and microstructure of the etched silicon surface were investigated using JEOL JSM 6360 LV (JEOL Ltd., Akishima-shi, Japan) scanning electron microscope (SEM) equipped with energy dispersive X-ray spectroscopy (EDS) and a FEI Inspect F-SEM (FEI, Hillsboro, OR, USA) equipped with FEG (field emission gun). The Fourier transform infrared spectrometry (FTIR) was performed with a Thermo Nicolet spectrometer (Thermo Fisher Scientific, Waltham, MA, USA). IR spectra were recorded at a resolution of 4 cm⁻¹ by averaging 32 scans.

The presence of Pd traces during the etching was performed using secondary ion mass spectroscopy (SIMS) 4E7.

Results and discussion

The morphology of the Si etched surface was examined by SEM. Figure 1a,b,c shows SEM images of silicon surface etched in V₂O₅ solution at different times, 30, 60, and 120 min. For an etching time of 30 min (Figure 1a), the formation of nanoporous silicon layer appears in the form of islands separated by large channels, which resembles the electrochemically etched n-type silicon surface in HF/ethanol solution [12,13]. Increasing the etching time to 60 min (Figure 1b) leads to the formation of pyramidal or pillar structures with a flat summit. These pillar structures become rounded facet with a curved and sharp summit at an etching time of 120 min (Figure 1c). The height of the pillar measured by SEM is about 10 μm.

The obtained morphologies are found to be different from those generally observed with other oxidants [14,15], such as H₂O₂, K₂CrO₇, HNO₃, where different structures
have been obtained. For example, it was shown that etching of Si substrate in HF/H₂O₂ solutions leads to the formation of nanoporous and macroporous structures. The stain etching can produce both PSi and silicon nanowires (SiNW). The hydrogen terminated surface (H-Si) is important in the formation of both structures. The obtained morphologies are probably due to the particularity of using V₂O₅ which dissolves in HF to form VO₂⁺ (one hole is injected) which is reduced to VO²⁺ during PSi nanostructure formation. Indeed, the discovery of stain etching with V₂O₅ as oxidant by Kolasinski et al. [16] has provided a unique opportunity to study the stoichiometry of stain etching. Moreover, Kolasinski et al. [16] have explained the simultaneous doubling of VO₂⁺ consumption and 50% reduction in H₂ evolution by etching according to the following reaction:

\[
\text{Si} + 2\text{HF} + 2\text{HF}_2^+ + \text{VO}_2^+ \rightarrow \text{SiF}_2^- + \text{H}_2 + \text{VO}^{2+} + \text{H}_2\text{O} + e^-_{cb}
\]  

This reaction is accompanied by a charge-balancing counter reaction

\[
\text{VO}_2^+ + 2\text{H}^+ + e^-_{cb} \rightarrow \text{VO}^{2+} + \text{H}_2\text{O}
\]  

Figure 2a,b,c,d shows SEM images of Pd-coated Si subjected to etching solution for 30, 60, 90, and 120 min where the presence of circular pores on the surface is revealed. In addition, it is shown that the pore size evolves with the increase of etching time (Figure 2). Pore diameter increases rapidly for an etching time of 60 min, to reach a value of about 7 μm, and then increases slowly for etching time higher than 120 min. A value of 9 μm is measured for an etching time of 350 min as depicted in Figure 3. This behavior could indicate that the catalyst effect of Pd becomes less predominant as the chemical reaction time increases.

Indeed, the amount of Pd deposited on silicon surface reacts and dissolves in the etching solution leading to the diminishing of the concentration of holes on the silicon surface and then of the etching rate. This chemical etching leads to the etching in the interior of macro pores and formation of pyramidal structures. The increase in the etching time favors the formation of new structures inside the macro pores. As shown in the inset of Figure 2d, as the Pd effect disappears, the pore wall will be broken by the etching solution and new pyramidal structures appear.
The widening of the pores could be due to different reasons: (1) Si without metal is etched as well due to the diffusion of injected holes from the etching front to the side of the pore wall [17,18], and (2) Si without Pd metal coverage is etched slowly in the etchant [19].

Elemental analysis by EDS of Pd-coated Si surface subjected to the etching solution for 120 min (Figure 4) revealed that the Si surface is mainly composed of Si and a bit of oxygen, probably due to oxidation of the Si sample. One can note the absence of Pd on Si surface, indicating its consummation during the chemical etching. SIMS characterization performed on Pd/Si etched samples for different times (30 to 120 min) confirms the absence of palladium traces into the samples at 9 μm from the top of the surface, as shown in Figure 5. We can observe that the Pd signal is under the detection limit (10⁴ to 10⁵ c/s).

Etching time effect on the rate was examined by measuring the etch depth as a function of etching time in the range of 0 to 360 min. The etch depth was also compared with that measured by SEM.

Figure 6a shows the measured etch rate of Si samples etched in HF/V₂O₅ (0.12 M). It is seen that the etch rate increases rapidly with etching time, reaches a maximum at 60 min, and then decreases. The diminishing in the etch rate is due to the decrease of vanadium concentration in the solution.

Figure 6b corresponds to the etching of Pd-coated Si in solution as previously mentioned. It shows that the etch depth first increases linearly with time, reaches a maximum at about 20 min, then decreases to attain a minimum etch rate value for 40 min, and finally increases to stabilize at an etching time of 70 min. This behavior is interesting since it confirms that the etching of Pd-coated Si samples in HF/V₂O₅ occurs in two steps. In the first step, the metal-assisted chemical etching is predominant (Pd catalyst effect), and in the second step, the stain etch is involved (Pd is consumed in the etching).

FTIR measurements of Si and Pd/Si etched samples in HF/V₂O₅ solution are discussed in the following section. Etched Si samples in HF/V₂O₅ reveal the presence of hydride (SiHₓ) species on the surface (Figure 7a). A shouldering, associated with monohydride (SiH), appears at 2,081 cm⁻¹. The spectrum is dominated by pronounced peaks at 2,113 cm⁻¹, associated with dihydride (SiH₂) species, which becomes larger over etching time.

For the Pd/Si etched films (Figure 7b), the surfaces are H-terminated. It has been shown that the Si-H bonds
evolve with etching time. At an etching time of 15 min, the formation of Si-H bonds appears. We note also the presence of peaks at 2,188, 2,215, 2,245 cm\(^{-1}\) indicating that O atoms are inserted in Si-H surface bonds during the second part of the etching. The formation of these SiH\(_x\)O\(_y\) bands is observed at longer etching time, probably caused by oxidation induced by water during the PSi formation.

The decrease of PL intensity with increasing time is probably due to the reduction of nanostructure (nanocrystallites) density.

The reflective properties of the samples etched at different conditions in HF/V\(_2\)O\(_5\) solution are shown in Figure 9a,b. It is shown that Si etched for 120 min presents a low mean reflectance of 2.21% in the wavelength range of 400 to 1,000 nm. This can be explained by the uniform pyramidal structures shown in Figure 1c. The structures obtained at short etching time show a high reflectance probably due to the densification of the structure. Whereas for the Pd/Si samples etched in the same

It is interesting to notice that the peak intensity can be easily controlled by the etching time.

Figure 8 shows photoluminescence (PL) spectra dependence on etching time. We observe a red shift and a decrease of PL intensity with increasing etching time. The maximum PL intensity is found for an etching time of 30 min at 590 nm, while the etched layer for 120 min exhibits luminescence with a peak at 610 nm.
conditions, as previously mentioned, the mean reflectance increases as the etching time increases. A mean reflectance value of 4.33%, in the wavelength range of 400 to 1,000 nm, is obtained for an etching time of 15 min. The results demonstrate that these etching methods reduce the surface reflectance to values less than 4.33%. Thus, they can be used in the fabrication process of photovoltaic cells.

Conclusions
The results show that Si etching in HF/V₂O₅ solution induces the formation of pyramidal or pillar structures which evolve with etching time. The presence of metal catalyst (Pd) does not only accelerate the etch rate but also creates pyramidal structures within macropores whose diameter increases with etching time till the total consummation and disappearance of the Pd film. A low mean reflectance value of 2.21% in the wavelength range
Acknowledgements

Part of this work has been performed at the NanoFacility Piemonte, INRIM, a laboratory supported by Compagnia di San Paolo in the framework of the EU Project NaS-ERA. The authors thank Sabrina Sam from CRTSE for her contribution to the revisions of the FTIR characterization and in designing the manuscript. The authors also acknowledge Lakhdar Guer bouze from the Centre de Recherche Nucléaire d’Alger for photoluminescence characterization. The authors thank Hamid Menari for reflectance characterization.

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Received: 30 April 2014 Accepted: 26 August 2014

Published: 11 September 2014

References

1. Schmidt V, Bel H, Senz S, Karg S, Riess W, Gösele U: Realization of a silicon nanowire vertical surround-gate field-effect transistor. Small 2005, 1:285.

2. Goldberger J, Hochbaum AI, Fan R, Yang P: Controlled growth of Si nanowire arrays for device integration. Nano Lett 2006, 6:973.

3. Tian B, Zheng X, Kempa TJ, Fang Y, Yu N, Yu G, Huang J, Lieber CM: Coaxial silicon nanowires as solar cells and nanoelectronic power sources. Nature 2007, 449:865.

4. Peng KQ, Xu Y, Wu Y, Yan Y, Lee ST, Zhu J: Aligned single-crystalline Si nanowire arrays for photovoltaic applications. Small 2005, 1:11062.

5. Sivakov V, Andra G, Gawlik A, Berger A, P lentz J, Falk F, Christiansen SH: Silicon nanowire-based solar cells on glass: synthesis, optical properties, and cell parameters. Nano Lett 2009, 9:1549.

6. Chan CK, Peng HL, Liu G, McIvor K, Zhang XF, Huggins RA, Cui Y: High-performance lithium-ion battery anodes using silicon nanowires. Nat Nanotechnol 2008, 3:31.

7. Peng K, Jie J, Zhang W, Lee ST: Silicon nanowires for rechargeable lithium-ion battery anodes. Appl Phys Lett 2008, 93:033105.

8. Cui Y, Wei Q, Park H, Lieber CM: Nanowire nanosensors for highly sensitive and selective detection of biological and chemical species. Science 2001, 293:1289.

9. Vásárhelyi E, Szalágyi F, Petr ık P, Horváth ZE, Lohner T, Fried M, Jalsovszky G: Porous silicon formation by stain etching. Thin Solid Films 2001, 388:295.

10. Kolasinski KW, Yadlowski J: Stain etching of silicon with V2O5. Phys Status Solidi C 2011, 8(6):1749–1753.

11. Beale MJ, Benjamin JD, Uren MJ, Chew NG, Cullis AG: The formation of porous silicon by chemical stain etches. J Crystal Growth 1986, 80(1):408–414.

12. Beale MJ, Chew NG, Uren MJ, Cullis AG, Benjamin JD: Microstructure and formation mechanism of porous silicon. Appl Phys Lett 1985, 46:86.

13. Hadjersi T, Gabouze N, Yamamoto N, Sakamaki K, Takai H: Photoluminescence from photochemically etched highly resistive silicon. Thin Solid Films 2004, 459:249–253.

14. Hadjersi T, Gabouze N, Yamamoto N, Banazouzi C, Cheraga H: Blue luminescence from porous layers produced by metal-assisted chemical etching on low-doped silicon Vacuum 2005, 80:366–370.

15. Kolasinski KW, Barclay W: The stoichiometry of electrosilicon etching in solutions of V2O5 and HF. Angew Chem Int Ed 2013, 52:3551–3554.

16. Tsigino K, Matsumura M: Helical nanoholes bored in silicon by wet chemical etching using platinum nanoparticles as catalyst. Electrochem Solid St Lett 2005, 8:193.

17. Chartier C, Bastide S, Levy-Clement C: Metal-assisted chemical etching of silicon in HF-H2O2. Electrochim Acta 2008, 53:5509–5516.

18. Huang JP, Shimizu T, Senn S, Zhang Z, Zhang XX, Lee W, Geyer N, Gösele U: Ordered arrays of vertically aligned [110] silicon nanowires by suppressing the crystallographically preferred <100> etching direction. Nano Lett 2009, 9:2519.

19. Lehmann V: Electrochemistry of Silicon: Instrumentation, Science, Materials, and Applications. Weinheim, Germany: Wiley-VCH; 2002.

Cite this article as: Ayat et al. Formation of nanostructured silicon surfaces by stain etching. Nanoscale Research Letters 2014 9:482.