Multiferroic bismuth ferrite: Perturbed angular correlation studies on its ferroic $\alpha$-$\beta$ phase transition

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Work of numerous research groups has shown different outcomes of studies of the transition from the ferroelectric $\alpha$-phase to the high temperature $\beta$-phase of the multiferroic, magnetoelectric perovskite Bismuth Ferrite (BiFeO$_3$ or BFO). Using the perturbed angular correlation (PAC) method with $^{111m}$Cd as the probe nucleus, the $\alpha$ to $\beta$ phase transition was characterized. The phase transition temperature, the change of the crystal structure, and its parameters were supervised with measurements at different temperatures using a six detector PAC setup to observe the $\gamma$$\cdot$$\gamma$ decay of the $^{111m}$Cd probe nucleus. The temperature dependence of the hyperfine parameters shows a change in coordination of the probe ion, which substitutes for the bismuth site, forecasting the phase transition to $\beta$-BFO by either increasing disorder or formation of a polype type transition structure. A visible drop of the quadrupole frequency $\omega_0$ at a temperature of about $T_c \approx 820^\circ$C indicates the $\alpha$-$\beta$ phase transition. For a given crystal symmetry, the DFT-calculations yield a specific local symmetry and electric field gradient value of the probe ion. The $Pbnm$ ($\beta$-BFO) crystal symmetry yields calculated local electric field gradients, which very well match our experimental results. The assumption of other crystal symmetries results in significantly different computed local environments not corresponding to the experiment.

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I. INTRODUCTION

Significant work on multiferroic materials started to take place through Soviet research in the late 1950s. Generally, a multiferroic can be described as a material that exhibits a combination of at least two of the following ferroic forms of ordering, namely, ferromagnetism (which here includes every kind of magnetic appearance with a macroscopic magnetic moment $M$ plus antiferromagnetism), ferroelectricity, and ferroelasticity, in the same phase [1]. Great focus lies on materials which show a combination of ferromagnetic and ferroelectric properties, as the simultaneous appearance of these forms opens new ways of exploiting material properties like electrically controlled magnetization and vice versa [2].

Wide fields of application are promised in spintronic and data storage devices as well as in sensoric and actuatoric systems [3]. In this aspect, among all discovered multiferroics, Bismuth Ferrite (also referred to as BFO), owing to its high antiferromagnetic Néel temperature of $T_N \approx 370^\circ$C and high ferroelectric Curie temperature of $T_c \approx 825^\circ$C [4], turned out to be a very promising candidate because it could be trained to exhibit a magnetoelectric effect even at room temperature. The $\alpha$-phase of BFO appears in a rhombohedral structure with the cell parameters $a = 5.58$ Å and $c = 13.87$ Å in the hexagonal settings at room temperature. It is a quasicubic cantz perovskite structure of the LiNbO$_3$ type [4]. The $\alpha$-phase exists up to 820°C where BFO shows a phase transition to the $\beta$-phase, which is accompanied by a sudden contraction of the unit cell volume. Its character still is the subject of numerous discussions and is mentioned to appear in cubic, trigonal, and even tetragonal structures, as can be seen in Fig. 1 [5–11]. Eventually, Arnold et al. [9] suggested the $\beta$-phase to appear in the orthorhombic $Pbnm$ space group with the cell parameters $a = 5.61$ Å, $b = 5.64$ Å, and $c = 7.97$ Å. To characterize the $\alpha$-$\beta$ phase transition in BFO, we applied

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II. EXPERIMENTAL DETAILS

The polycrystalline BiFeO₃ samples were synthesized at the University of Duisburg-Essen from a stoichiometric mixture of finely powdered bismuth oxide (Bi₂O₃, 99.9% Acros Organics) and iron oxide (Fe₂O₃, 99.99% Alfa Aesar). Following a calcination process for 3 hours at 820 °C, the powder was pressed to pellets and sintered for six hours under air atmosphere again at 820 °C. The samples were shipped to the ISOLDE/CERN facility, where, to perform the PAC measurements, the Cadmium isotope ¹¹¹mCd was implanted at the ISOLDE GLM beamline with an accelerator voltage of 30 keV. According to simulations with the SRIM software package [21], the implantation depth distribution shows a maximum at around 116 Å. The PAC measurements were done immediately after implantation in the facilities of the ISOLDE Solid State Physics Collaboration [13] using a fully digital six detector spectrometer [22]. The furnace with an isolated sample holder allowed measurements at all desired temperature points. All measurements were performed at temperatures reached from below. No PAC measurement was done on cooling due to the time limits in the experiment. PAC, being a hyperfine interaction technique, allows the investigation of the electromagnetic fields interacting with a nuclear probe located in the material [23,24]. The probe nucleus in its excited state decays to its ground state via the sensitive intermediate state through a γ-γ cascade decay. Through an angular correlation between the first and the second γ-ray, distinguished through their energies and detection times, hyperfine interactions between the probe nuclei and their surrounding atomic structure can be monitored. A fixed detector, which detects a first quantum, defines a reference axis, conventionally designated as the z-axis. A number of nuclei with their initial spin J, aligned to this direction is selected. This nuclear alignment leads to an anisotropic radiation of γ₂ with regard to γ₁ [25]. If the probe nucleus quadrupole momentum is disturbed through an electric field gradient (EFG) while being in the intermediate state, its spin J will experience a torque which forces a spin-precession around the before-referenced z-axis. Caused by this precession, the...
population of the intermediate state sublevels changes with time, which again affects the spacial anisotropy of the $\gamma_1$-$\gamma_2$ correlation. An electric field gradient is caused by charge distributions which deviate from an ideal spherically symmetric arrangement around the probe atom. It is defined as the second spacial derivative of the electrostatic potential. Through a diagonalization of the interaction Hamiltonian, rotating the tensor into its principal component coordinate system and the use of the Laplace equation, the EFG can be described by its largest component $V_{zz}$ and the asymmetry parameter $\eta = \frac{V_{zz} - V_{yy}}{V_{zz}}$.

only. The quadrupole interaction frequency is calculated through

$$\omega_\eta = \frac{eQ V_{zz}}{4(I - 1)\hbar}$$

where $e$ is the electron charge, $Q$ the quadrupole moment of the intermediate state of the probe nucleus, $I$ its angular momentum, and $\hbar$ the reduced Planck’s constant. For an anisotropy coefficient $\eta$ of zero, the lowest detectable substrate-transition frequency is

$$\omega_0 = k \omega_\eta$$

with $k = 6$ for half-integer values of spin momentum, like the $I = 5/2$ level of our probe isotope $^{111}$mCd. The higher-ordered frequencies are calculated through $\omega_n = n \omega_0$. If $\eta$ is different from zero, the transition frequencies $\omega_n$ are functions of $\omega_\eta$ and $\eta$ itself:

$$\omega_1 = 2\sqrt{3} \alpha \omega_\eta \sin \left(\frac{1}{3} \arccos(\beta)\right)$$

$$\omega_2 = 2\sqrt{3} \alpha \omega_\eta \sin \left(\frac{1}{3} \left[\pi - \arccos(\beta)\right]\right)$$

$$\omega_3 = 2\sqrt{3} \alpha \omega_\eta \sin \left(\frac{1}{3} \left[\pi + \arccos(\beta)\right]\right)$$

with

$$\alpha = \sqrt{\frac{28(3 + \eta^2)}{3}}$$

and

$$\beta = \frac{80(1 - \eta^2)}{\alpha^3}.$$  

These transition frequencies represent the transitions between the substate levels created through the hyperfine splitting of the intermediate state. For a $I = 5/2$ state and a pure quadrupole moment, the intermediate state is split into three sublevels $E_Q(M)$, resulting in three transition frequencies $\omega_n$ visible in the perturbation function [Eqs. (4) to (6)]. The normalized values of the sublevel energy states $E_Q(M)$ as well as of the transition frequencies $\omega_n$ and their change with rising asymmetry parameter $\eta$ are shown in Figs. 2 and 3. Following Gerdau [26], the theoretical perturbation factor for the fit of polycrystalline samples is described by

$$G_{kk} (\eta, t) = \sum_{n=0}^{3} s_{kn}(\eta) \cos[\omega_n(\eta, V_{zz})t] \exp(-\delta\omega_n t),$$

FIG. 2. Change of sublevel energies $E_Q(M)$ with increasing asymmetry parameter. Reference values only valid for $I = 5/2$.

with $\delta$ being the Lorentzian EFG distribution width and $s_{kn}(\eta)$ being weight functions depending on the asymmetry parameter $\eta$. Exact formulas for the weight functions $s_{kn}(\eta)$ for isotopes with an intermediate state of $I = 5/2$ can also be found in Friedsam [27]. The perturbation function of the $\gamma$-$\gamma$ angular correlation is finally represented by the anisotropy function

$$R(t) = A_{kk} G_{kk}(t)$$

$$= A_{kk} \sum_{n=0}^{3} s_{kn}(\eta) \cos[\omega_n(\eta, V_{zz})t] \exp(-\delta\omega_n t),$$

with $A_{kk}$ being correlation coefficients depending on the nuclear cascade and $G_{kk}$ containing all information about the EFG [23,27–29].

III. RESULTS AND DISCUSSION

A. Simulation results

As mentioned before, the published data on the $\alpha$-$\beta$ phase transition temperature and the structure of the $\beta$-phase
TABLE I. Simulated values of $V_{zz}$ and $\Delta H$ as well as the calculated value of $\omega_0$. The range of the results of $\omega_0$ is caused by the uncertainty of $Q$.

| System          | $V_{zz}^{DFT}$ [bohr$^2$] | $\eta^{DFT}$ [1] | $\Delta H^{DFT}$ [eV] | $\omega_0^{DFT}$ [Mrad/s] |
|-----------------|--------------------------|-----------------|-----------------------|---------------------------|
| $\alpha$-BFO (FM/AFM) | Cd @ Bi                  | 5.72/5.37       | 0/0                   | 8.4/2.6                   | 80.3–86.8/75.4–81.5       |
|                 | Cd @ Fe                  | 5.46/4.07       | 0/0                   | 8.8/5.5                   | 76.7–82.9/57.1–61.8       |
| $\beta$-BFO (FM/AFM) | Cd @ Bi                  | −7.51/−6.88     | 0.39/0.26             | 4.2/1                     | 105.4–113.9/96.57–104.41  |
|                 | Cd @ Fe                  | 2.37/8.43       | 0.87/0.18             | 5.4/2.2                   | 33.3–35.9/118.33–127.93   |
| Bi$_2$Fe$_4$O$_9$ (FM) | Cd @ Bi                  | 9.99            | 0.12                  | −                        | 140.22–151.61             |
|                 | Cd @ Fe                  | 7.99            | 0.84                  | −                        | 112.15–121.25             |

Contradict. The crystal structure of the perovskite BFO offers two different sites for the implanted $^{111m}$Cd probe atom, namely the A-site, substituting the bismuth site and the B-site, substituting iron. To determine the most favorable substitution site as well as to set benchmarks for verification of the PAC results, DFT simulations were done using the Vienna Ab Initio Simulation Package (VASP) program [30]. The employed approach was the projector augmented wave method with a general gradient approximation using the 5$d^{10}6s^26p^6$ valence configuration for Bi, $2s^22p^6$ configuration for Fe, $3p^63d^84s^2$ configuration for O, and the $4d^{10}5s^2$ configuration for Cd. For the $\alpha$-phase ($Z = 6$) calculations, a $2 \times 2 \times 1$ supercell in the hexagonal setting, with chemical formulas CdBi$_{15}$Fe$_{16}$O$_{48}$ for Cd substituting Bi and Bi$_2$CdFe$_{23}$O$_{72}$ for Cd substituting Fe was used. We used a $k$-point grid of $4 \times 4 \times 3$ for the calculation. The $\beta$-phase ($Z = 4$) calculations were also done within a $2 \times 2 \times 1$ supercell with the formulas CdBi$_{15}$Fe$_{16}$O$_{48}$ for Cd substituting the Bi-site and Bi$_{16}$CdFe$_{18}$O$_{48}$ for Cd substituting the Fe-site with a $k$-point grid of $5 \times 5 \times 4$. Bi$_2$Fe$_2$O$_9$ ($Z = 2$) is a secondary phase detected in the x-ray diffraction measurements with space group $Pbam$ and lattice parameters $a = 7.9826$ Å, $b = 8.49636$ Å, and $c = 6.04298$ Å. With this cell the distance between Cd impurities ($c \approx 6$ Å) may be small, so we used a $2 \times 1 \times 2$ supercell where the smallest distance between Cd impurities increases to the lattice parameter $b \approx 8.5$ Å ($k$-point grid of $3 \times 2 \times 4$). The $k$-point densities were similar considering the supercell sizes. For all calculations the energy cutoff was 450 eV. The self-consistent calculations stopped for energy differences between two steps smaller than $10^{-5}$ eV. Other precision settings were set by the VASP tag PREC=ACCURATE. The atomic positions are optimized to minimize the forces below a threshold of 0.01 eV/Å. The $U$-parameters in use were a PBE approximation with Coulomb repulsion $U_{eff} = 3$ eV applied to Fe-$d$ electrons [31]. The electric field gradient $V_{zz}$, the anisotropy parameter $\eta$, and the formation energy $\Delta H$ were simulated for Cd substituting the Bi, as well as the Fe sites. The quadrupole interaction frequency $\omega_0$ was then calculated with the use of formulas 2, 3, and 4 with the simulated values of $V_{zz}$, $\eta$, and $Q = 0.641(25)$ barn [32]. Even though BFO is paramagnetic at our measurement temperatures, the paramagnetic order is very difficult to simulate. Temperature was not taken into account in the simulations. The paraelectric phase was not simulated with molecular dynamics as the computational cost would have been out of proportion considering the outcome. As a workaround, all simulations were done for ferromagnetic (FM) and $G$-type antiferromagnetic (AFM) magnetic spin order to determine the EFG dependence on magnetic ordering in general. Simulation results and the calculated values of $\omega_0$ of the $\alpha$-phase, the $\beta$-phase and an additional ternary phase, Bi$_2$Fe$_2$O$_9$, which will be explained in Sec. III D, can be found in Table I. As the results of the $\alpha$-, and the $\beta$-phases show, the formation energy $\Delta H$ is lower in both the FM and the AFM order in the simulated system of Cd substituting the A-site. Thus for the dynamics explained in this work, a Cd substitution on the Bi site is energetically more favorable than a substitution on the Fe site, which is backed by the findings of Gebhardt [33]. We also calculated the local effect of two other structures which have been proposed for the $\beta$ phase, considering Cd substitutional at Bi. For the $R3c$ structure, $V_{zz} = 18.87 \times 10^{21}$ V/m$^2$ and $\eta = 0$ were obtained at the Cd site, substitutional at Bi. The corresponding $\omega_0 \approx 250$ Mrad/s is much higher than experimental values, while $\eta = 0$ also does not agree with the high $\eta$ measured in the $\beta$ phase. For the P2$_1$/m structure, a large supercell calculation ($3 \times 2 \times 3$ of the original P2$_1$/m) was used. The obtained EFG at Cd is $V_{zz} = −24.8 \times 10^{21}$ V/m$^2$, $\eta = 0.37$. In this case we used a smaller energy cutoff of 400 eV, and the forces are smaller than 4 mRy/bohr radius), a slightly less accurate calculation, due to the large computational time required. Nevertheless, the result should already have qualitatively converged for $V_{zz}$, and the corresponding $\omega_0 \approx 350$ Mrad/s is very far from the experimental value, allowing us to conclude that this is not the measured configuration.

B. PAC results

Figure 4 shows the experimental $R(t)$ functions as well as fast fourier transformation (FFT) data with fits at measurement temperatures between 800° C and 850° C. The FFT analysis is helpful to visualize the $R(t)$ functions through the transition frequencies $\omega_0$ and their ratio. The connection to the asymmetry parameter $\eta$ (see Fig. 3) of the containing EFGs is directly visible.

As mentioned in Secs. III A and III C, the Cd probe ion substitutes the Bi site within the BFO sample material, leading to sensitivity to changes in its nearest surrounding, which is set up by oxygen atoms forming coordination polyhedra around the Bi(Cd) atom. The change from a sixfold coordination in the $\alpha$-phase to an eightfold coordination in the $\beta$-phase results from a change of the Bi-site symmetry. An ideal, nondistorted perovskite structure with point group $Pm3m$ shows a 12-fold coordination on the $A$-site and a 6-fold coordination on the $B$-site. In the case of BFO though, the perovskite structure is
FIG. 4. TDPAC spectra with fitting functions in red and FFT showing the measured EFGs.

distorted, which changes the bond lengths of the coordinating atoms drastically. Regarding that, the coordination numbers mentioned here describe the number of nearest coordinating oxygen atoms which influence the environment of the cation most. The Bi ion in the \( \alpha \)-phase is located on a threefold rotation axis so symmetric by 3-fold rotation (site symmetry \( 3 \), trigonal point group). On the other hand, the site symmetry of the Bi position in the \( \beta \)-phase is \( m \), which refers to a monoclinic point group, meaning that the Bi ion is located on a mirror reflection plane and so symmetric by mirror reflection. This change in symmetry is accompanied by a change in average coordination lengths from \( d_{\text{Bi-O}} = 2.42 \, \text{Å} \) [34] to \( d_{\text{Bi-O}} = 2.64 \, \text{Å} \) [9], both visible in the PAC results as the following chapter will illustrate. The visualization of the single EFGs in Fig. 4 already hints towards the changes happening during the phase transition from the \( R3c \) to the \( Pmmm \) phase as it shows a clear assignment of EFG1 to the \( \alpha \)-phase and of EFG2 to the \( \beta \)-phase. The data acquisition as well as the fitting and the calculation of the experimental EFG were done with the GFIT19 software package, a modified version of the NNFIT software package [35,36]. Hyperfine parameters of all fits were calculated and can be found in Table II. The higher number of near neighboring oxygen atoms in the surrounding of the cadmium probe atom in eight-fold coordination (\( \beta \)-phase) leads to an increased quadrupole interaction frequency with respect to the 6-fold coordination in the \( \alpha \)-phase which can be seen in Fig. 5. Up to a measurement temperature of 804°C, only the lower frequency EFG (\( \omega_0 \sim 86 \, \text{Mrad/s} \)) is present. Compared with the simulation results listed in Table I, we associate this field gradient (EFG1) to \( \alpha \)-BFO in ferromagnetic order with Cd@Bi. The measured asymmetry parameter \( \eta \) also matches the simulated value reasonably well. As is visible on the FFT of EFG1 in Fig. 4, the low value of \( \eta \) close to zero leads to a \( \omega_1 : \omega_2 : \omega_3 \) ratio of 1:2:3 (compare to Fig. 3). Starting at 807°C, also the high frequency field gradient (EFG2, \( \beta \)-phase) is detected. As seen in Fig. 6, the values of \( \omega_0 \) assigned to the \( \alpha \)-phase are practically constant throughout the temperature increase. On the other hand, the quadrupole frequency of EFG2 which is assigned to the \( \beta \)-phase shows a decrease from \( \omega_0 = 123.67 \, \text{Mrad/s} \) at 818°C to \( \omega_0 = 110.05 \, \text{Mrad/s} \) at 822°C. This can also be seen in the FFT functions in Fig. 4 as EFG2 clearly changes between 818°C and 822°C. This feature is the indication for the phase transition. The discordant values of \( \omega_0 \) of EFG2 right before that decrease indicate the following: With increasing temperature and the approaching of the phase transition temperature, local coordination disordering of the Bi-site, which below 820°C is still in its 6-fold \( \alpha \)-phase coordination, appears. This means that a pseudo \( m \)-symmetry coordination, which actually belongs to the not yet existing \( \beta \)-phase, starts to appear on the 3-fold-symmetry site of the Bi ion. This disordered coordination distorts the EFG and leads to the high values of \( \omega_0 \). The second parameter to distinguish between the \( \alpha \) and the \( \beta \) phases is the asymmetry parameter \( \eta \) of EFG2.
TABLE II. Experimental PAC values of hyperfine parameters.

| Temp. [°C] | α-phase | β-phase |
|------------|---------|---------|
|            | ω₀\(_{\text{PAC}}\) [Mrad/s] | η\(_{\text{PAC}}\) [1] | δ\(_{\text{PAC}}\) [%] | % | ω₀\(_{\text{PAC}}\) [Mrad/s] | η\(_{\text{PAC}}\) [1] | δ\(_{\text{PAC}}\) [%] | % |
| 500        | 106(6)  | 0.44(7) | 30(7)  | 100 |                           |         |         |   |
| 603        | 91(3)   | 0.43(6) | 19(2)  | 100 |                           |         |         |   |
| 650        | 88.5(3) | 0.14(2) | 4.6(4) | 100 |                           |         |         |   |
| 703        | 87(1)   | 0.23(3) | 10(1)  | 100 |                           |         |         |   |
| 776        | 86.8(5) | 0.16(2) | 10(1)  | 100 |                           |         |         |   |
| 800        | 86.6(3) | 0.10(2) | 5.3(3) | 100 |                           |         |         |   |
| 804        | 87.3(5) | 0.16(4) | 10.3(6)| 100 |                           |         |         |   |
| 807        | 86.5(2) | 0.08(1) | 2.0(3) | 78(5) | 122(0) | 0.92(1) | 0.2(7) | 22(2)|
| 810        | 86.8(2) | 0.02(2) | 2.6(3) | 79(9) | 120.54(0) | 0.99(4) | 4(1) | 21(3)|
| 814        | 86.5(2) | 0.07(2) | 3.3(3) | 65(6) | 121.14(0) | 0.99(2) | 5.6(7) | 35(2)|
| 818        | 86.3(2) | 0.08(1) | 2.2(2) | 61(2) | 123.67(0) | 0.90(3) | 10(1) | 39(2)|
| 822        | 85.4(5) | 0.16(1) | 2.5(1) | 50(6) | 110.05(0) | 0.83(0) | 0.7(3) | 50(6)|
| 826        | 88.0(1) | 0.15(2) | 4.4(9) | 49(5) | 109.66(0) | 0.83(1) | 0.03(6) | 51(5)|
| 829        |         |         |         |     | 109.4(1) | 0.8(0)  | 2.2(2) | 100 |
| 831        |         |         |         |     | 109.2(1) | 0.84(0) | 0.2(1) | 100 |
| 840        |         |         |         |     | 106.6(2) | 0.86(0) | 0.6(2) | 100 |
| 850        |         |         |         |     | 105.3(3) | 0.86(0) | 0.5(2) | 100 |

FIG. 6. Temperature dependence of the quadrupole interaction frequency ω₀. The blue area shows the rhombohedral α-phase region and the red area the orthorhombic β-phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

Thus, the crystal structure itself changes likely due to longer range interactions. The disordering of the Bi-sites and the following changes in coordination are an independent effect, which are only visible locally. The visible changes in damping δ, even though they are small, have a significant meaning. EFG1 by comparison shows a peak in damping at 804°C. This feature indicates the emergence of the β-phase EFG, caused by the mentioned pseudo m-symmetry coordinations, which is already present at the next measurement point at 807°C. The increasing δ-value between 807°C and 818°C in EFG2 was already addressed before. At 822°C, the damping again drops to a value of δ\(_{822}\) = 0.68(2)%). This indicates that the phase transition is completed. From now on, the Bi ion sits on the mentioned site with m-symmetry in an eightfold coordination. EFG1 exists up to a temperature of 826°C, where the last parameter η. As clearly seen in Fig. 7, the α-phase appears in a rather symmetric setting (η\(_{\alpha}\) ∼ 0.1) compared to the β-phase (η\(_{\beta}\) ∼ 0.9). There is a slight change of the asymmetry parameter in EFG2 between 814°C and 822°C as the value drops from η\(_{814}\) = 0.996 to η\(_{822}\) = 0.828. This dynamic is linked to the coordination disordering described before. The surrounding of the probe atom thus experiences a noticeable structural change in this temperature region. An indication for these dynamics are the slightly increasing values of damping δ in the β-EFG between 807°C and 818°C (Fig. 8), as with the increasing number of disordered Bi-coordination sites the EFG distribution width broadens. The damping δ, shown in Fig. 8, is generally low at all measurements, indicating a rather narrow EFG distribution and thus allows an accurate analysis. It also shows that the change in EFG due to fluctuations is low.

FIG. 7. Temperature dependence of the asymmetry parameter η. The blue area shows the rhombohedral α-phase region and the red area the orthorhombic β-phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.
FIG. 8. Temperature dependence of the damping parameter \( \delta \).
The blue area shows the rhombohedral \( \alpha \)-phase region and the red area the orthorhombic \( \beta \)-phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

The remaining pseudocoordinates with 3-fold-symmetry vanish. As can be seen in Fig. 9, the fraction of EFG2 is increasing with increasing temperature while the fraction of EFG1 decreases. This, once again, underlines the dynamics of the coordination change as the \( ^{111}\text{m} \text{Cd} \) probe ion forecasts the phase transition. An alternative explanation which is not in contradiction with our measurement results was given by Prosandeev et al. [38] with a polytype phase that acts as an intermediate structure between the \( R3c \) \( \alpha \)-phase and the \( Pbnm \) \( \beta \)-phase. This putative transition structure is designated “nano-twinned” in the work of Prosandeev et al. PAC data, as mentioned, allow a very specific, local picture of the closest environment of the probe atom. Thus, the resulting polytype structure, with a period length of six lattice constants along [001] cannot be measured as a unique, independent EFG but can be interpreted through the coincident appearance of EFG1 and EFG2. In contrast to the predictions of Prosandeev et al., the observed transition structure is stable only in a 19 K as opposed to a 250 K region. Considering this, the phase transition can be explained as a combination of two independent phase transitions following each other. The visible transition range around the \( \alpha - \beta \) phase transition in BFO to our knowledge is a novel result. This range is thus not only a response to the substitutional Cd cations but also to global crystal symmetry change: On the one hand, crystal defects, such as the Cd substitution in BFO, lead to strong distortions in their closest vicinity which also might induce a delayed or accelerated phase transition, at least on a local point of view. The local defect induced broadening of the transition range can only be measured with PAC and not with XRD. On the other hand, the XRD data of pure BFO show a hysteresis around \( T_c \) (see below), which is smaller but with a width of around 8 K accounts for roughly 45% of the transition range width measured with PAC. In consideration of the mentioned details, we assume that there is a hysteresis of the long-range order parameter. The increase of the frequency fraction attributed to the \( \beta \)-phase in Fig. 9 can be interpreted as a phase transition of the second order. Such a second-order phase transition could either mean an increase of the fraction of the polytypes predicted by Prosandeev et al. or a continuous decrease of the periodicity of the repetition period of the stacking, i.e., an increase in \( \beta \)-BFO like coordination.

C. Probe site assignment

Site assignment for tracer ions is sometimes a difficult task. We here collect arguments for \( A \) or \( B \) site substitution. DFT simulations, which are only strictly valid at a temperature of 0 K, might to some extent be uncertain at our measurement temperatures. A comparison of the ionic radii of the involved elements strongly suggests a substitution of the Bi site through the Cd ion. Depending on the coordination number (CN), the ionic radii of all ions vary as summarized in Table III. As well as the Fe cations are assumed to be in a 6-fold coordination in the \( \alpha \)-phase, with their effective ionic radii of 1.02 Å and 0.55 Å (low spin)/0.645 Å (high spin), respectively. Thus, the value of a 6-fold coordination of Cd\(^{2+}\) was taken for comparison [9]. The 6-fold effective ion radius of Cd with 0.95 Å is much closer to the radius of the Bi ion than to the Fe ion, which is the most important indication for the substitution.

![Table III. Effective ionic radii of Cd\(^{2+}\), Bi\(^{3+}\), and Fe\(^{3+}\) ions in Å for different coordination numbers CN [39].](image)

| CN | Ion | \( \text{Cd}^{2+} \) | \( \text{Bi}^{3+} \) | \( \text{Fe}^{3+} \) | \( \text{In}^{3+} \) | \( \text{Mn}^{3+} \) |
|----|-----|---------|---------|---------|---------|---------|
| IV | 0.84 | –       | 0.49    | –       | –       |
| V  | 0.87 | 0.99    | –       | –       | 0.58    |
| VI | 0.95 | 1.02    | 0.55 (LS) | 0.79 | 0.58 (LS) |
|    |      |         | 0.645 (HS) |       | 0.65 (HS) |
| VII| 1    | –       | –       | 0.923   | –       |
| VIII| 1.07 | 1.11    | –       | –       | –       |
| XII| 1.31 | –       | –       | –       | –       |
at the A-site [39]. A substitution of the Fe cation would necessarily lead to a significant distortion of the unit cell, possibly followed by an asymmetry parameter different from zero. A counterargument would be that ferroelectricity in BiFeO$_3$ is mainly due to the lone pair polarization of the Bi ion. Thus, Cd substituting Bi must experience the local effect of changes in polarization in its field gradient. So, the EFG should change also before the curie point itself ($\alpha$ to $\beta$ phase transition) is reached. Landau theory predicts a moderate change of the order parameter just beneath such a strong first-order phase transition [40]. So unlike in second-order phase transitions, where a large change in $V_{zz}$ would have to precede the phase transition, the jump in order parameter at the first-order phase transition temperature allows for only moderate changes just beneath $T_c$. We observe practically no change in $V_{zz}$ beneath the phase transition temperature and assign this to the very limited temperature range we investigate. In this range no change is within the error bar of analysis with a “moderate” change which would be a few percent only as we know from other data on BiFeO$_3$. The $^{111}$In probe site substitution was also a point of discussion in rare-earth orthochromite ceramics. Comparisons with $^{111}$In($^{111}$Cd) PAC measurements done on rare-earth orthoferrites lead to the assumption that in the particular case of SmCrO$_3$ the $^{111}$In/Cd probe also occupies the A-site [41]. The $^{111}$In($^{111}$Cd) probe used to examine the rare-earth orthoferrites (RFeO$_3$, REO) occupies both sites, leading to a low frequency EFG describing the Fe-occupancy and a high frequency EFG describing the R-occupancy [42]. As seen in Table III, $^{111}$In has a significantly smaller ionic radius (0.79 Å in 6-fold coordination) than $^{111}$In/Cd, which possibly leads to an occupation of both sites. Rearick et al. [42] also mentioned that the $^{111}$In probe primarily substitutes into the rare-earth sites in the heavier REOs. Though bismuth is not a rare-earth element, an $\alpha$-site occupation is more likely as Bi is heavier than all the rare earth elements. Further literature affirming the A-site substitution are the works of Bellakki [43], which addressed Cd-doped BFO and Ishaq et al. [44], which treats Cd-Mn doped BFO. Bellakki states that Cd substitutes only the Bi cation. Ishaq et al. described that Cd substitutes the Bi site whereas Mn substitutes the Fe site. Reconsidering the ionic radii of Cd and Mn (Table III) this is expected as the value of Mn is significantly closer to the one of Fe than the ionic radius of Cd which is similar to Bi. Sato et al. also describe $^{111}$Cd as substituting the R and A sites in doped perovskite manganese oxides ($R_1$-$A_x$MnO$_3$) [45]. As in our work only one EFG is assigned to the $\alpha$-phase, A-site occupancy is what we conclude from these arguments.

D. XRD results

The powder diffraction measurements were performed in air using a PANalytical X’Pert Pro MPD $\theta$-$\theta$ diffractometer equipped with an Anton Paar HTK 1200 high temperature chamber and an X’celerator 128 channel detector. Intensity data were collected in the $2\theta = 20^\circ$–$70^\circ$ range with a scan length of 2.122$^\circ$ with 55.2 s exposure per scan length resulting in overall measurement times of about 45 minutes. Data were converted into step size bins of 0.0084$^\circ$. This measurement routine was used to determine the overall phase composition of the samples at different temperature points around the presumed Curie temperature as suggested by the PAC measurements. Sweep measurements covering the $2\theta$-range between 30$^\circ$ and 35$^\circ$ were done to determine the exact phase transition temperature. Samples were ground and the powder was placed on platinum sample holders to avoid reactions with standard sample holders such as Si single crystals, quartz or Macor. Figure 10 shows XRD diffractograms of BFOs $R3c$ and $Pbnm$ structure, respectively. The intensities are normalized to the highest intensity.

FIG. 10. Temperature dependence of x-ray diffractograms measured between $20^\circ$–$70^\circ$ (graphs are only shown up to $60^\circ$ as no information gain is achieved from reflections above $60^\circ$) at increasing temperatures around the phase transition temperature. The top and bottom graphs show calculated diffractograms of BFOs $R3c$ and $Pbnm$ structure, respectively. The intensities are normalized to the highest intensity.

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$2\theta = 31.5^\circ$, which is most likely caused by a temperature gradient. Rietveld refinement results in an $\alpha$-BFO fraction of ca. 8%. The additional reflections appearing in the diffractograms are either Pt reflections from the sample holder (marked with *) or of the known decomposition products Bi$_2$Fe$_4$O$_9$ (#) and Bi$_{25}$FeO$_{39}$ (~) as BFO is metastable at elevated temperatures [46]. The two reflections labelled with x were not identified but could result from sample contamination during the measurements. As mentioned, the phase transition was also analyzed through sweep measurements in a small angle range, determined to ascertain the exact phase transition temperature as well as the hysteresis around the transition point. A suitable reflection for this examination was found to be the 104 reflection and also an attenuated intensity of the $\alpha$-phase fraction. Again, the heating cycle is depicted in red and the cooling cycle in blue. Refinement of the $\alpha$-phase close to the phase transition temperature [47] could not be confirmed. However, we did not observe any anomaly in the thermal expansion of the $\alpha$-phase close to the phase transition temperature. The temperature difference of 2 K between the phase transition temperature observed in these sweep measurements in the heating cycle (820 °C) and the measurements covering the angle between 20° and 70° (818 °C) can be explained through uncertainties of the temperature measurement in the high temperature chamber during the different single experiments. Reflections of the ternary mullite Bi$_2$Fe$_4$O$_9$ and the selenite Bi$_{25}$FeO$_{39}$ phases are visible in all measured diffractograms. The reaction from the secondary phases back to BiFeO$_3$ as proposed by Selbach et al. [47] could not be confirmed. However, Fig. 13 shows that at 840 °C the 013 reflection of Bi$_{25}$FeO$_{39}$ is

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Bi$_2$Fe$_4$O$_9$ with temperature. The color gradient corresponds to the temperature change.

measurements, easing the volatility of Bi and thus the building of powdered samples in comparison to bulk samples used during the PAC caused through the vigorously larger surface area of powdered which were not exposed to a prior PAC measurement might be of those phases. Visible ternary phase fractions in samples XRD measurements which leads to an increase of fractions could be explained through the second heating process at the measurements, which were conducted after the PAC measurements, The appearance of the ternary phases in the XRD measure-ments, which were done after the PAC results including phase fractions of Bi$_2$Fe$_4$O$_9$ with the calculated parameters. However, no phase fractions of the mullite phase were present in the PAC spectra. The appearance of the ternary phases in the XRD measurements, which were conducted after the PAC measurements, could be explained through the second heating process at the XRD measurements which leads to an increase of fractions of those phases. Visible ternary phase fractions in samples which were not exposed to a prior PAC measurement might be caused through the vigorously larger surface area of powdered samples in comparison to bulk samples used during the PAC measurements, easing the volatility of Bi and thus the building of the decomposition products.

IV. CONCLUSION

The ferro to paraelectric phase transition of multiferroic BiFeO$_3$ was investigated through two fundamentally different experimental methods. PAC, which is an extremely sensitive method probing the sample on an atomic scale and XRD. The results of both experimental methods agree with a phase transition from rhombohedral $\alpha$-BFO in $R3c$ setting to orthorhombic $\beta$-BFO with its $Pbnm$ space group. The assumption of the Bi-site substitution by the $^{111}$mCd probe is confirmed by crystal chemical assignment (ion radii) as well as by our DFT and PAC results. Following the PAC results, the first-order phase transition is forecast by the Cd probe ion as the preceding symmetry change of Bi-coordination is sensed. The phase transition point at 820°C is visible through a drop in quadrupole interaction frequency $\omega_Q$ which indicates the site symmetry change of the Bi position. The discordant values of $\omega_Q$ and $\eta$ within the temperature interval of 807°C and 818°C could be explained by the gradual formation of a polytype transition structure or by growing disorder. In the former case, the alpha-beta phase transition would actually be a combination of a second order followed by a first-order phase transition. The first could not be evidenced in XRD. Thus, it is shown that both methods are complementary. The transition range in PAC and thermal hysteresis measured with XRD suggests a change of the long-range order parameter around the phase transition point. A broadening of the transition range at the phase transition as a result of the disturbed local environment around the probe ion itself is visible through the wider range at the PAC measurements compared to the XRD results. The visibility of the mentioned possible dynamics during the phase transition in the results over a temperature range of 19 K again underlines the precision of PAC measurements at high temperatures, which transcends the performances of other nuclear measurement methods like Mössbauer or nuclear magnetic resonance spectroscopy in this temperature range. This is also a very fundamental result as we have proof of the local change of the order parameter caused by defects, which was often assumed in literature before [4]. Following the XRD results, the phase transition happens at a transition temperature of 818°C as this is the only temperature point where both phases appear simulta-neously. Indications of a polytype transition structure could not be found in our XRD data but its existence cannot be excluded. The appearance of the ternary phases Bi$_2$Fe$_4$O$_9$ and Bi$_{25}$FeO$_{39}$ was observed. They only arise in powdered samples of high surface area as were used for XRD and are not contained in any PAC data.

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