Optical methods of silicon nanoparticle diagnostics for applications in biomedicine

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Abstract. The method of Raman scattering spectroscopy was used to study various silicon nanostructures (nanowires, mesoporous nanoparticles, crystalline and laser-ablated nanoparticles) dispersed in aqueous medium. The obtained results indicate different dissolution rate for silicon nanoparticles of different sizes and morphology in water that can be used for their potential biomedical applications. © 2019 Journal of Biomedical Photonics & Engineering.

Keywords: silicon nanocrystals; nanowires; porous silicon; nanocrystalline silicon; laser ablated nanoparticles; Raman scattering; biodegradation.

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1 Introduction
In recent years, various types of nanostructured forms of crystalline silicon (Si) have been intensively studied for application in technology and biomedicine, including for the diagnosis and therapy of various diseases [1, 2]. Porous Si [3, 4–7], silicon nanowires [8], and nanoparticles (NPs) obtained from them [9], as well as laser-ablated Si NPs [2, 10] and nanoparticles obtained from porous c-Si [11], have a great potential of unique physical and chemical properties for application both in tumor therapy [1, 2, 4, 9] and as contrast agents for bioimaging [8, 10, 11].

At present, several methods for obtaining powders and stable aqueous suspensions of Si NPs are known [1–6, 9, 10]. Photoluminescence (PL) of laser-ablated Si NPs, which is explained by radiative recombination of excitons excited in silicon nanocrystals with dimensions of 3–5 nm, was successfully applied in the study of the NP penetration into living cells [10]. It was found that Si NPs can be effectively accumulated in cancer cells and therefore can be used for diagnostics and therapy of cancer [2]. It was shown that the processes of accumulation and excretion of Si NPs in biological systems can be evaluated by the Raman spectroscopy [9].

In our work, we use the Raman scattering spectroscopy to study an effect of water exposure on the structural properties of crystalline Si nanowires (SiNWs), mesoporous silicon (meso-PSi), silicon nanocrystals (nc-Si) obtained by plasma-chemical synthesis or by the method of femtosecond laser ablation of c-Si targets in water (abl-Si).

2 Experimental technique
Samples of SiNWs were formed by metal-stimulated chemical etching (MSCE) of single-crystal silicon wafers (c-Si) of p-type conductivity with resistivity of 1–10 Ohm·cm and surface orientation (100) by using the standard two step approach (see e.g. Ref. [8]).

Meso-PSi samples were formed using the standard method of electrochemical etching plates c-Si p-type conductivity with surface orientation (100) and resistivity of 1–5 mOhm·cm mixture of hydrofluoric acid and ethanol (HF50%: C2H5OH=1:1) at a current density 60 mA/cm² for 1 hour.

After that, meso-PSi films were separated from the silicon substrate by a short-term increase in the current density up to 600 mA/cm². Aqueous suspensions of nanoparticles were obtained as a result of mechanical grinding of meso-PSi films in the planetary ball mill FRITSCH “Pulverisette 7” for 9 minutes at a rotation speed of 800 rpm and a diameter of 5 mm zirconium oxide balls used (see, for example, review [3]). Powders of nc-Si were obtained by the method of plasma-
chemical synthesis with the use of pure (99.9%) microcrystalline Si as the feedstock (see, for example, Ref. [1]).

The samples of SiNWs were placed in a 4 cm$^3$ cuvette filled with deionized water or saline (0.9% NaCl in H$_2$O) and held for a number of days. As a starting material, a sample with lateral sample sizes of about 1×1 cm and a nanowire length of about 10 µm was used. To study the dissolution of PF, aqueous suspensions of meso-PSi, nc-Si and abl-Si with a concentration of about 1 g/l in a volume of 10 ml were placed in dialysis bags with pore sizes of 6–10 kDa and kept from 1 to 24 hours in a large volume (5 liters) of deionized water at room temperature. Then at intervals 1, 2, 3, 6, 12, 24 hours some part of the suspension was collected, were planted on the stainless steel substrate, air-dried for 10–15 min and then examined by using the Raman scattering. All experiments were conducted at room temperature.

The Raman spectra of the obtained samples were taken on the DFS-52 spectrometer, which is located in the laboratory of the Department of low temperatures and superconductivity of MSU, in 90-degree scattering geometry. The receiver served working in the photon counting mode a cooled photomultiplier tube PMT-79. As a source of exciting radiation, an argon laser with wavelengths of 488 and 514.5 nm and a maximum intensity of not more than 100 W/cm$^2$ was used. The HORIBA Jobin Yvon LabRAM HR Visible micro-Raman spectrometer was used to measure Raman scattering spectra with excitation at a wavelength of 632.8 nm or an argon laser with a wavelength of 488 nm. The Raman spectra were also measured at the “Horiba Jobin Yvon” MicroRaman LabRAM HR Visible facility when excited by HeNe laser radiation with a wavelength of 632.8 nm or an argon laser with a wavelength of 488 nm. The signal was recorded in the backscattering geometry at normal incidence at room temperature in the air.

3 Results and discussion

Raman spectra of silicon nanowires with average cross-sectional dimensions of 20–100 nm before and after exposure to water or normal saline for 2 weeks are shown in Fig. 1. In the spectrum, there is a line at 520.5 cm$^{-1}$ with width about 4 cm$^{-1}$, corresponding to the scattering on optical phonons of the c-Si crystal lattice. The high intensity of the line in comparison with single crystals of c-Si indicates the penetration of exciting radiation into the SiNWs layer followed by multiple scattering. The forms of the Raman scattering line in both c-Si and SiNWs samples are well described by the Lorentz function:

$$I_{nc}(ω) = \frac{B \exp\left(-\frac{1}{4} q^2 L^2\right) q^2}{(ω- ω_c)^2 + \frac{1}{2} Δω_c^2} dq, \quad (2)$$

where $ω_c = 520.5$ cm$^{-1}$ frequency phonons, the parameter $L = d/a_0$, $d$ is the diameter, $a_0 = 0.543$ nm is the lattice constant of c-Si, $q$ – the wave vector of the Phonon, is expressed in units ( $2π/a_0$ ), and $Δω_c = ω_c (1-0.18q^2)$ the ratio of the dispersion of the phonon in c-Si [13], $B$ is a constant.

Eq. (2) describes the confinement effect (phonon restriction effect) in spherical Si nanocrystals with sizes 1-10 nm [12], while for Raman scattering with large nanoparticles the expression (1) can be used.
defects be explained by a higher concentration of structural defects. In this case, abl-Si NPs exhibit significantly lower dissolution rate (Fig. 5), compared with NPs obtained by plasma-chemical method that can be explained by a higher concentration of structural defects in the latter.

The Raman spectrum of amorphous silicon (a-Si) is described by the following expression [13]:

\[ I_A(\omega) = A \times \exp \left( -\frac{(\omega - \omega_c)^2}{2\delta^2} \right), \]  

where \( A \) is a constant, \( \omega_c = 480 \text{ cm}^{-1} \) and \( \delta = \Gamma(2\sqrt{2}\ln 2)^{1/2} \) are the peak position width, respectively.

Assuming that the Raman scattering signal of the sample under study consists of inclusions of all fractions (phases) the formula can be presented as follows:

\[ I_{\text{sum}}(\omega) = I_c(\omega) + I_{\text{NC}}(\omega) + I_s(\omega). \]  

The volume fraction of the nc-Si phase can be calculated as the ratio of the corresponding integrated intensity to the total intensity of the Raman signal:

\[ f_{\text{NC}} = \frac{\int I_{\text{NC}}(\omega) d\omega}{\int I_{\text{sum}}(\omega) d\omega}. \]  

The Raman spectra of meso-PSi analyzed by using Eqs. (1)–(4) show that the average size of Si nanocrystals is about 5 nm and it does not practically change after the storage of the sample in aqueous solution. However, the contribution of the nanocrystalline phase becomes smaller and the signal of a-Si fraction increased (see Fig. 3). The latter indicates an additional disordering of the crystal lattice of NP Si due to their partial dissolution in water in accordance with the findings of Ref. [9].

Investigation of Raman scattering from nonporous nc-Si with sizes from 5 to 100 nm, obtained by plasma chemical (Fig. 3) and laser ablation methods (Fig. 4), indicates partial dissolution of nanoparticles, resulting in a decrease in their total number and average size of nanocrystals. In this case, abl-Si NPs exhibit significantly lower dissolution rate (Fig. 5), compared with NPs obtained by plasma-chemical method that can be explained by a higher concentration of structural defects in the latter.

The obtained results indicate different dissolution rates of Si nanostructures prepared by wet chemical route and physical methods. Thus, SiNWs, due to their high stability in water, can be used as elements of biosensors. NPs of mesoporous silicon can be...

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**Fig. 2** Raman spectra for the initial meso-PSi (1) sample and after dialysis in water for 24 h (2).

**Fig. 3** Raman spectra of nc-Si samples before (1) and after exposure in water for 1 (2) and 24 (3) hours.

**Fig. 4** Raman spectra of abl-Si samples before and after dialysis in water.

**Fig. 5** Raman spectra of aqueous suspension of abl-Si source and c-Si substrate.
recommended for the purpose of express-diagnostics and therapy, while NPs produced by plasma-chemical synthesis and laser ablation of c-Si in water can be used as agents for short-time and prolonged simultaneous diagnosis and therapy (theranostics) of cancer diseases.

Disclosures
All authors declare that there is no conflict of interests in this paper.

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