Compressive strength of calcium phosphate cements prepared using different initial setting temperatures

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The effect of initial setting temperature on the compressive strength of calcium phosphate cements prepared using Ca₃(PO₄)₂O and CaHPO₄ was examined. The strength of the initial setting bodies increased with increasing temperatures: the strength of the hydroxyapatite following initial setting of the bodies treated at higher temperatures and the delayed hydration reaction following simulated body heated at 37°C was 3 times higher than that heated at 100°C. This was due to the rapid formation of hydroxyapatite following initial setting of the bodies treated at higher temperatures and the delayed hydration reaction following soaking in simulated body fluid.

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1. Introduction

Various calcium phosphate cements (CPCs) have been studied for application as bone substitutes. Bone reconstruction using CPC paste, which has been shaped nearly into bone defect dimensions prior to implantation to body, is known as one of useful treatments in plastic surgery. In general, the surgical sites need to be sutured immediately after implementation of the CPC pastes to prevent any infection. During such intervention, inflammation around soft tissues owing to leakage of non-reacted raw CPC powders from the defect site can be observed. Additionally, there is a serious concern about the fracture of CPC with insufficient mechanical strength when CPC setting is not enough. CPC is required to set in the desired time following formation of the CPC paste and maintain its shape during such operation.

It has been reported that additives, such as organic acids, in the CPC pastes can accelerate hydration reactions and reduce the setting time. Shorter setting times can potentially lead to an unsuccessful operation, owing to rapid transformation processes. Additionally, treatment at high temperatures can shorten the setting time of CPC paste. To date, the effects of short setting times achieved by including heating process on the mechanical properties of the resulting CPCs are unclear.

In the present work, CPC pastes were heated at different temperatures during the initial setting process. The influences of temperature on the compressive strength and operating hydration reactions within the CPC setting bodies were examined.

2. Experimental procedure

Dicalcium phosphate anhydrous (DCPA; CaHPO₄, Taihei Chemical Industrial, Nara, Japan; mean particle size of 8.0 μm) was milled with distilled water for 24 h using an alumina ball mill, and dried at 120°C for 20 h following our previous work. The resulting DCPA sample featured a mean particle size of 0.6 μm. Tetracalcium phosphate [TeCP; Ca₄(PO₄)₂O] was prepared as follows: the mixture consisting of dicalcium phosphate dihydrous (CaHPO₄·2H₂O; Kanto Kagaku, Tokyo, Japan) and calcium carbonate (CaCO₃; Kanto Kagaku, Tokyo, Japan) at a weight ratio of 1.75:1 was fired at 1550°C for 10 h and then quenched at room temperature. The resulting matter was crushed to prepare powders with an average particle size of 200 μm. The raw CPC powders were prepared by mechanically mixing 67.5 g TeCP and 32.5 g DCPA using an automatic mortar. A 48.7 wt % dextran sulfate sodium aqueous solution (Meito Sangyo, Aichi, Japan) was used as mixing liquid. The raw CPC powders were mixed with the mixing liquid at a liquid-to-powder mass (g) ratio of 0.26:1 for 1 min at room temperature to prepare the CPC paste. The resulting CPC paste was filled in a stainless steel ring (10 mm in inner diameter, 5 mm in height) and flattened on its top surface to measure the setting time. A Vickers needle (300 g in weight, 1 mm² in cross-sectional area) was set on the top surface of the filled CPC paste at 37, 60, 80, or 100°C. The setting time was determined as the point at which no indent was formed by the needle (n = 5). The CPC paste was molded into disks of 6 mm in diameter and 12 mm in height, and heated at 37, 60, 80, or 100°C for the initial setting process. Some reports have been shown the evaluation on the compressive strength of CPC after soaking in simulated body fluid (SBF). In the present work, after 5 min of heating, the CPC setting bodies were soaked in SBF at 37°C for 24 h.

The compressive strength of the CPC setting bodies before and after the SBF soaking was evaluated on an Universal Testing Machine (Autograph AGS-5kND; Shimadzu, Kyoto, Japan). Following compressive strength measurements, the samples were immediately soaked in acetone for 1 h and dried under vacuum for 24 h. The crystalline phase of the dry ground fine powders was identified by X-ray diffraction (XRD; RU-200; Rigaku,
Tokyo, Japan). The relative intensity of HA phase was calculated from XRD patterns using the following equation:

\[
\text{Relative intensity of HA} \times 100 = \frac{I_{\text{HA}}}{I_{\text{DCPA}} + I_{\text{TeCP}} + I_{\text{HA}}}
\]

- \( I_{\text{DCPA}} \): Intensity of DCPA (020), \( 2\theta = 26.4^\circ \)
- \( I_{\text{TeCP}} \): Intensity of TeCP (200), \( 2\theta = 25.3^\circ \)
- \( I_{\text{HA}} \): Intensity of HA (002), \( 2\theta = 25.9^\circ \)

The microstructure of the raw CPC powders and fracture surfaces of the dry samples were observed by scanning electron microscopy (SEM; S-2500; Hitachi, Tokyo, Japan).

3. Results and discussion

In the present work, the relationship between the compressive strength and setting reaction at initial setting temperature was investigated for clinical application.

Table 1 shows the setting time of the CPC paste at each initial setting temperature. The time shortened with increasing the initial setting temperature. The setting time of the paste heated at 80 or 100°C was drastically short, that is, within 2 min.

Figure 1 shows the compressive strength of the CPC setting bodies. The samples prepared at higher temperatures possessed higher strength following initial setting for 5 min when compared with those prepared at lower temperatures. In contrast, samples prepared at higher temperatures followed by SBF soaking featured lower strength properties relative to those prepared at lower temperatures followed by SBF soaking. No collapse of CPC occurred after soaking in SBF. The CPC setting bodies heated at 37°C showed the highest compressive strength, ~50 MPa, following SBF soaking; prior to soaking, they featured a compressive strength of ~2 MPa. Conversely, the CPC setting bodies heated at 100°C displayed compressive strengths of ~14 and ~18 MPa before and after SBF soaking, respectively.

Figure 2 shows the XRD patterns of the CPC setting bodies. The setting bodies heated at 37°C displayed almost no changes in the crystalline phases following initial setting. Contrarily, hydroxyapatite (HA) peaks appeared following SBF soaking accompanied with a decrease in the intensity of the TeCP and DCPA peaks. In contrast, the setting bodies heated at 100°C displayed HA peaks and the intensity of the TeCP and DCPA peaks decreased following initial setting. Small changes in the intensity of HA, TeCP, and DCPA peaks were observed following SBF soaking. The sample heated at 37°C followed by SBF soaking showed a higher peak intensity of the HA phase and lower peak intensities of the TeCP and DCPA phases, when compared with counterpart sample heated at 100°C.

Figure 3 shows SEM images of the CPC powders and the fracture surfaces of the CPC setting bodies. DCPA particles (~0.5 μm in size) were clearly observed in the CPC powders. For the setting bodies heated at 37°C, although DCPA particles were observed after 5 min, following SBF soaking, they almost entirely disappeared. The CPC setting bodies heated at 37°C followed by SBF soaking showed a higher peak intensity of the HA phase and lower peak intensities of the TeCP and DCPA phases, when compared with counterpart sample heated at 100°C.
disappeared and numerous submicrometer-sized particles that are believed to be HA (based on XRD analysis) were observed [Fig. 2(c)], as consistent with previous report. In contrast, for the CPC setting bodies heated at 100°C, DCPA particles and HA were observed in the sample after 5 min. Almost no changes were observed following SBF soaking.

Figure 4 shows the compressive strength of CPC setting bodies as a function of the relative intensity of HA. The initial setting temperature significantly influenced the compressive strength and HA formation. This is believed to originate from the difference in the setting behavior and operating hydration reactions in CPC. The initial hydration reaction rate in CPC increased with increasing initial setting temperatures that contributed to the increase in the compressive strength. In contrast, the CPC sample following SBF soaking for 24 h displayed a reduced compressive strength and delayed HA formation with increasing initial setting temperatures. The difference in HA formation was observed in the XRD and SEM analyses. It has been reported that the diametral compressive strengths of the CPC setting bodies heated at 100°C, DCPA particles and HA were observed in the sample after 5 min. Almost no changes were observed following SBF soaking.

The compressive strength of CPC following initial setting increased with increasing initial setting temperatures, whereas that of CPC following SBF soaking at 37°C for 24 h was not drastically enhanced with increasing initial setting temperatures—hydration reactions were delayed. This was likely because of the difference in the crystallinity of the formed HA and the change in the solubility of DCPA and TeCP following heating.

Control of the hydration reaction by varying the initial setting temperature is expected to prevent inflammation owing to CPC collapse after implantation.

4. Conclusion

The compressive strength of CPC following initial setting increased with increasing initial setting temperatures, whereas that of CPC following SBF soaking at 37°C for 24 h was not drastically enhanced with increasing initial setting temperatures—hydration reactions were delayed. This was likely because of the difference in the crystallinity of the formed HA and the change in the solubility of DCPA and TeCP during initial setting.

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