Luminescent Lead Halide Ionic Liquids for High-Spatial-Resolution Fast Neutron Imaging

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ABSTRACT: The fast neutron imaging technique with recoil proton detection harbors significant potential for imaging of thick, large-scale objects containing high-Z elements. However, the challenge to find efficient fast neutron scintillators with high spatial resolution is ongoing. The list of requirements for such scintillators is long and demanding: a proton-rich, scattering-free material combining high light yield with the absence of light reabsorption. To meet these challenges, we look for a suitable material among a rising class of 0D organic–inorganic Pb(II) halide hybrids. The use of large organic cations, e.g., trihexyltetradecylphosphonium, results in room-temperature ionic liquids that combine highly Stokes-shifted (up to 1.7 eV), reabsorption-free, and efficient emission (photoluminescence quantum yield up to 60%) from molecularly small and dense (PbX₃, molar fraction up to 0.33) emitting centers. We investigate the optical properties of the resulting ionic liquids and showcase their utility as fast neutron imaging scintillators. Concomitantly with good light yield, such fast-neutron scintillators exhibit both higher spatial resolution and lower γ-ray sensitivity compared with commercial ZnS:Cu-based screens.

KEYWORDS: ionic liquid, fast neutrons, scintillation, imaging, lead halides

INTRODUCTION

As opposed to thermal neutrons with relatively low penetration depth or high-energy X-rays with insufficient low-Z element contrast, fast neutrons (1–15 MeV) are the tool of choice for imaging thick objects containing both high- and low-Z elements.¹ The primary method for fast neutron detection is based on elastic scattering of neutrons by nuclei, generating recoil nuclei (typically protons) with very small penetration depths (up to tens of micrometers) in the detector material. The kinetic energy of these recoil nuclei is then deposited as ionized charge carriers in the detector material, which can excite a scintillator to emit visible-range photons detectable by conventional imaging devices such as a CCD camera. However, the potential of this method depends on the exploration of efficient scintillator materials. The perfect scintillator for this indirect detection scheme should be a scattering-free, proton-rich, and high-light-yield material with a dense concentration of (preferably atomically small) fast emitting centers.

As the state-of-art fast neutron scintillators face their own challenges,² researchers start to divert their course toward ns²-metal halides,³ a rapidly expanding class of optoelectronic materials with promising applications in high-energy detection.³ Vast structural versatility within this class results in a plethora of materials, which can be roughly divided into semiconductors (3D corner-sharing ABX₃ and related 2D structures)⁵,⁶ and electronically and structurally lower-dimensional compounds (non-corner-sharing 2D, 1D, and isolated 0D).⁷,⁸ For instance, colloidal ABX₃ nanocrystals (NCs) dispersed in a proton-rich medium (organic solvents) make for a suitable scintillation medium;¹ however, their strong self-absorption (small Stokes shift) limits achievable spatial resolution and light yield. Zero-dimensional metal halides, due to their molecular-like largely Stokes-shifted emission, are particularly successful in indirect detection schemes as X-ray scintillators.⁹⁻¹¹

Herein, we explore 0D Pb(II) metal halides as scintillators for fast neutron detection and imaging. Zero-dimensional Pb(II) halides are typically composed of isolated anionic PbX₃ centers (mono- or polynuclear)¹⁰,¹³ and inorganic (e.g., Cs⁺)¹⁴,¹⁵ or organic cations. The emission originates from electron and hole pairs trapped on metal halide species¹⁶ and can be tuned in the UV-to-green spectral range by halide (Cl or Br) and coordination number (3, 4, 6).¹⁷,¹⁸ The reabsorption-free quality of their emission renders a good...
prerequisite for high spatial resolution in fast neutron imaging. However, the majority of 0D materials are crystalline bulk ionic salts. Among approaches to achieve scattering-free formulations of ns² 0D materials are colloid NCs, which are thus far limited to fully inorganic compositions, such as 0D Cs₅PbBr₆ (nonluminescent), Cs₄SnX₆ (luminescent but air-sensitive), and others. Another strategy would be to prevent the crystallization of 0D salts, preferably at room temperature (RT). Such supercooled liquid materials are known as RT molten salts or ionic liquids (ILs). To form an IL, large organic ionic species are required: their bulkiness decreases the degree of ion association, which is reflected in a depressed melting point. These organic cations make also for a high loadings are viscous and pose inconveniences during their handling. It has to be noted that although there exists a study on the speciation in THTDPCI-PbCl₂ system, no optical properties of these species have been investigated. Both halide systems feature largely Stokes-shifted (200 nm) and broad molecular-like UV PL excitation (PLE); for the chloride, both PL and PLE are shifted toward higher energies (Figure 1c,d). Such optical properties are very characteristic of self-trapped 0D metal halide centers. Moreover, optical properties of ILs closely resemble previously studied species formed when lead(II) halides are dissolved in polar solvents (e.g., dimethylformamide, dimethyl sulfoxide, acetonitrile, polyethylene glycole) with an excess of halide anions provided by quaternary ammonium salts. Typically, the absorption maximum shifts toward lower energies with the increase of coordination number in the order PbX₄²⁻ > PbX₃⁻ > PbX₂⁻. However, we have found that the PL and PLE properties of the ILs change little in the compositional range from 0.09 ≤ χPbX₂ ≤ 0.25 (Table S1). The big difference in RT PL quantum yield (PLQY) between chloride and bromide ILs (60% vs 5%) can be attributed to more efficient phonon–exciton PL quenching for heavier Br, as has been observed for other 0D ns²-metal halides. On the basis of previous research, we tentatively assign the luminescence observed in samples with χPbX₂ ≤ 0.33 to [PbX₄²⁻] species. In extremely diluted samples (χPbX₂ = 0.0025), a blue-shifted PL peak around 420 nm starts to appear.

Figure 1. (a) Building blocks of haloplumbate(II) ILs: cationic species used in this study include TBP and THTDP phosphonium cations, while anionic speciation depends on the molar ratio between Pb and X (where X = Cl, Br). (b) Photo of the THTDPX-PbX₂ (X = Cl, Br) samples displaying typical PbX₄²⁻ emission under UV (365 nm) excitation. PL, PLE (dashed line) and absorption (solid line) spectra of the (c) 3THTDPCI-PbCl₂ and (d) 3THTDPBr-PbBr₂.

### RESULTS AND DISCUSSION

Halometallate ILs comprising metal halide anions and large organic cations (e.g., alkylmethylimidazolium (C₃mim)), tetraalkylammonium, tetraalkylphosphonium, etc.) are particularly common and span across most metals in the periodic table. The functionality of halometallate IL is typically defined by its anionic species, and catalytic, magnetic, and luminescent ILs have been reported. For the present study, we have chosen tetraalkylphosphonium cations, namely, tetrabutylphosphonium (TBP) and trihexyltetradeethylphosphonium (THTDP) to pair with lead(II) halide, yielding luminescent ILs. The choice was partially motivated by a previous study on chloroplumbate(II) IL speciation, which demonstrated that in the THTDPCI-PbCl₂ system the solidus line lies below −60 °C. The haloplumbate(II) ILs have been prepared by dissolving the required amount of PbX₂ in molten THTDPX (at RT) or TBPX (at T > 70 °C). Synthetic details can be found in the Methods section. Figure 1a depicts the main building blocks of the resulting ILs. According to a prior study, [PbX₄²⁻] species are present across the whole PbX₂ molar ratio range (0.55 ≥ χPbX₂ ≥ 0.05), while [PbX₃⁻] were only detected for χPbX₂ ≥ 0.33. Optical Properties of Lead(II) Halide ILs. Figure 1b demonstrates representative lead(II) chloride and bromide ILs with the THTDP⁺ cation, for a cation to lead(II) halide ratio of 3:1 or molar ratio χPbX₂ = 0.25. The maximum molar ratio χPbX₂ ≥ 0.33 is also achievable; however, the samples with such high loadings are viscous and pose inconveniences during their handling. It has to be noted that although there exists a study on the speciation in THTDPCI-PbCl₂ system, no optical properties of these species have been investigated. Both halide systems feature largely Stokes-shifted (200 nm) and broad molecular-like UV PL excitation (PLE); for the chloride, both PL and PLE are shifted toward higher energies (Figure 1c,d). Such optical properties are very characteristic of self-trapped 0D metal halide centers. In extremely diluted samples (χPbX₂ = 0.0025), a blue-shifted PL peak around 420 nm starts to appear.

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along with PbX$_4^{2-}$ PL, which can be attributed to species with higher coordination number, e.g., PbX$_6^{4-}$ (Figure S1).

**Fast Neutron Imaging with Lead(II) Halide ILs.** Proton-rich cations in the ILs are well suited for the indirect fast neutron detection scheme, depicted in Figure 2a. The high energy recoil proton, generated by the scattering of the fast neutron off the cation, generates electron–hole pairs, which in turn recombine on the emissive species with the generation of visible light photons. Fast neutron imaging in such a scheme can be easily realized by introducing a CCD camera that detects the emission behind the scintillator ILs, as depicted in Figure 2b. Lead(II) halide ILs combine attractive features of prospective scintillators for fast neutron imaging: $^{35-37}$ intimately mixed highly emissive molecular phosphors and long organic chains to provide sufficient stopping power for fast neutrons. $^3$ Furthermore, the transparency of these interdissolved materials eliminates the phosphor–plastic interface, which scatters light in the commercial ZnS:Cu(PP) phosphor screens, while their short-lived emission lifetimes (450 ns, Figure S2) also stand in stark contrast to the minutes-long afterglow effects observed in ZnS:Cu(PP) screens. $^2$ Most importantly, the large Stokes shifts of these emitters minimize self-absorption losses in light yield and eliminates photon recycling (reabsorption and emission) that would reduce spatial resolution.

**Light Output of Lead(II) Halide ILs in a Fast Neutron Imaging Experiment.** A representative composition THTDPCl-PbCl$_2$ (3:1 or $\chi_{PbX_2}=0.25$) (referred below as “IL”) and that diluted 1/5 with neat THTDPCl ("1/5-IL") were selected for fast neutron imaging measurements. These measurements were conducted at the NECTAR$^3$ beamline of the FRM-II research reactor (Garching, Germany). The details of the beamline and experimental setup have been described previously.$^3$ In short, the sample cuvettes are sealed inside a light-tight scintillation chamber with a CCD camera that detects the radioluminescence produced by the cuvettes upon fast neutron irradiation.

![Figure 2](image)

Figure 2. (a) Schematics of the processes involved in the recoil proton detection with ILs: (1) elastic neutron scattering on the H-rich cation with the generation of a recoil proton, (2) ionization of the recoil proton with subsequent generation of the electron–hole pairs in IL, and (3) recombination of the electron–hole pairs and emission of visible light photon. (b) Scheme of the indirect fast neutron imaging experiment with ILs as a scintillator.

![Figure 3](image)

Figure 3. (a) Radiograph of IL under fast neutron irradiation (300 s exposure) as compared with nanocrystal scintillators$^3$ and a commercial ZnS:Cu(PP) screen, used here as a reference. (b) Total light yield comparison of an ionic liquid with leading fast neutron NC scintillators$^3$ given as a percentage of the light yield obtained for the reference ZnS:Cu(PP) scintillator screen under identical conditions (300 s exposure, full concentration, 10-mm-thick cuvettes). (c) Light yield of concentrated and dilute IL vs vertical pixel position from the radiograph in panel (a), showing the light outcoupling at the cuvette edge and meniscus, as well as the imperfect mixing of the diluted sample leading to a higher PbCl$_2$ concentration at the bottom of the sample. (d) Concentration-dependent light yield of IL and selected NC scintillators.$^3$
A series of comparative radiographs of the ionic liquid samples and some colloidal NCs, along with a commercial ZnS:Cu(PP) screen, were measured under fast neutron exposure for times ranging from 30 to 300 s (e.g., Figure 3a). The results clearly demonstrate the proof-of-concept for fast neutron detection in ILs, with bright radioluminescence from these samples under exposure to a fast neutron beam (Figure 3a). The light yield of the ionic liquid is linear with increasing exposure time and reproducible across multiple series of such measurements (Figure S3), indicating the absence of degradation or long-lived afterglow effects. Note that imperfect mixing of the 1/5-diluted ionic liquid (1/5-IL) (done on-site) results in an uneven light yield across the height of the cuvette (side profile rotated for clarity, Figure 3c).

The total light yield is reported as a percentage of the total light yield of the reference ZnS:Cu(PP) scintillator under the same conditions, and we find that the light yields of the IL and 1/5-IL are 7.6% and 4.0% of the reference, respectively (Figure 3b). The light yields of IL and 1/5-IL are competitive with semiconductor NC colloids (e.g., 2.1 mass % for FAPbBr3 NCs) and is on par with the ideal loading observed for ZnS:Cu(PP), this may indicate that there is a proportion of lead(II) metal centers that are nonemissive and act as parasitic absorbers of charge carriers, degrading the PLQY and lowering the observed CCE.

The Conc-IL sample showed a discoloration due to absorption in the visible range (Figure S5), and accordingly the maximal light yield was achieved with the “IL” sample \(\chi_{PbX2} = 0.25\) (Figure S6). This indicates that the chemistry of the TTDPCl-PbCl2 system changes at high concentrations, with the addition of parasitic absorption that reduces the light yield under fast neutron irradiation. Nevertheless, the high light yield of this IL despite the possible presence of parasitic centers demonstrates the benefits of IL-based scintillators with significantly higher phosphor concentrations than that achievable with NCs.
Spatial Resolution of Lead(II) Halide ILs in a Fast Neutron Imaging Experiment. Turning to test the spatial resolution of the IL-based scintillator, a lead block (2.5 cm thick) is placed in the path of the fast neutron beam to provide a sharp edge to the transmitted beam, leading to light (unblocked) and dark (attenuated) regions in the scintillator (Figure 4a). The spatial resolution is then estimated by comparing the sharpness of the edge in the scintillator’s change in light yield across this region, as shown in the side profile of the cuvette (rotated for clarity, Figure 4b). Here, we utilize the 10–90% rise as a conservative numeric measure of the edge sharpness, normalizing the light and dark regions to 1 and 0 light output, respectively, and measuring the number of pixels needed to go from 0.1 to 0.9 (each pixel represents ~0.1 mm).

The brightest NCs (FAPbBr3 and CdSe/CdS) are included for comparison, and the undiluted IL exhibits significantly better spatial resolution than both NCs.

The edge measured with the IL is slightly sharper than that of the ZnS:CuPP reference (Figure 4c). The edge width of the IL is 19 pixels, far below the values of 52 and 29 pixels for FAPbBr3 and CdSe/CdS NCs, respectively, and even better than the ZnS:CuPP reference at 22 pixels (Figure 4d, Table S2). The 1/5-IL is lower still at 16 pixels, though the signal-to-noise ratio is reduced due to lower light yield.

The impressive spatial resolution observed for the IL highlights the critical role of self-transparency in obtaining high-spatial-resolution scintillators, whereas the NCs exhibit absorption and re-emission of scintillation light that obscures the sharp edge, clearly observed in FAPbBr3 with the emission tail extending throughout the dark region (Figure 4b). This result also demonstrates that the lack of a scattering phosphor–plastic interface can yield improvements even over the commercial standard ZnS:CuPP. Furthermore, the ZnS:CuPP phosphor screen (2.4 mm) is less than a quarter of the thickness of the IL cuvette (10 mm), because the resolution of such screens has been shown to decrease as thickness rises, forcing a trade-off between light yield and spatial resolution. In contrast, the large Stokes shift of the lead(II)-based IL exhibits excellent resolution despite thicker screens, allowing for scaled-up screens of arbitrary thickness to maximize the detector efficiency (and thereby light yield) without sacrificing spatial resolution.

An important criterion for a prospective fast neutron scintillator is the γ-ray sensitivity. For instance, the uranium fission reaction used at NECTAR to generate the fast neutron beam also produces a high γ-ray flux, which contaminates the fast neutron image if the phosphor is sensitive also to γ-rays. This is detrimental to the performance, as such effects must be counteracted by shielding the γ-ray contribution with heavy absorbers, thereby reducing the fast neutron flux. The blocking experiment provides a way to evaluate the contribution of γ-rays to the observed scintillation response: the lead block changes the light yield and has known neutron and γ-ray attenuation coefficients \(a_n\) and \(a_\gamma\). Thus, the ratio of the light yield in the light and dark regions of the IL, \(R_{\text{light}}\), can be used to derive the ratio \(R_{n/\gamma}\) of light yield induced by neutrons and γ-rays for a metal block with thickness \(d\):^40

\[
R_{n/\gamma} = \frac{e^{-a_\gamma d}}{1 - e^{-a_n d}}
\]

The γ-ray attenuation coefficient of lead was averaged with respect to the prompt γ-ray spectrum of the \(^{235}\text{U}\) γ-ray spectrum. The calculated neutron fission spectrum was used to estimate the fast neutron attenuation coefficient of the lead block.\(^43\) Note that this overestimates the neutron stopping power because the calculated average energy (1.35 MeV) is less than the actual average energy delivered at NECTAR (1.8 MeV),\(^38\) underestimating the neutron contribution to the light yield and thereby overestimating the γ-ray sensitivity.

The light yield ratio \(R_{n/\gamma}\) for the IL is 1.76(5), well above the reported value of 1.0–1.1 for ZnS:CuPP (as measured on the same beamline).\(^2\) This means that 64% of the scintillation light is derived from neutrons for the IL, while that of ZnS:CuPP is only 50–52%, enhancing the neutron-specific light yield of the IL to 9.7% of the ZnS:CuPP reference. Dilution further reduces the γ-ray sensitivity, as the 1/5-IL exhibits an \(R_{n/\gamma}\) of 1.97(5). While these values are not so high as the ratios obtained previously for the NCs (2.2–4.1),\(^40\) they are promising values considering the higher density of the concentrated IL relative to the dilute NCs. For reference, over 14 cm of Pb shielding is needed to lower the γ-ray contribution to 1% of the signal due to the low \(R_{n/\gamma}\) of ZnS:CuPP, while an \(R_{n/\gamma}\) of 1.7 reduces this to only 9.6 cm of Pb.\(^41\) This permits substantially higher fluxes to be used and drastically shortens collection times; to put this improvement in context, eliminating 4.4 cm of Pb shielding is equivalent to eliminating 1.8 of the 2.5-cm-thick Pb blocks used here to provide the contrast in Figure 4a. This allows for a much higher fast neutron flux and benefits tomography applications where reconstruction tasks are too calculation-heavy to remove the γ-ray-induced “background” signal.

CONCLUSIONS

In conclusion, this work demonstrates the proof-of-concept for IL-based fast neutron scintillators and provides guidelines to identify promising IL-metal halide compositions. Pb(II) halide ILs display a compelling set of characteristics required for fast neutron imaging: they are scattering-free, include proton-rich cations, and permit a high density of the emitting centers. These qualities, together with high PLQY (60%) and large Stokes shift (up to 1.7 eV), result in a spatial resolution superior to the established ZnS:Cu scintillators. Future work should build upon these advances by improving the PLQY of ILs and achieving a higher concentration of emissive species.

METHODS

Materials. Trihexyltetradecylphosphonium chloride (THTDPCI, >95%, IoLiTec), trihexyltetradecylphosphonium bromide (THTDPBr, >95%, IoLiTec), tetrabutylphosphonium chloride (TBPCl, 95%, ABCR), tetrabutylphosphonium bromide (TBBr, ≥98%, Fluka), lead(II) chloride (PbCl2, 99.999%, metal basis, ABCR), and lead(II) bromide (PbBr2, 99.999%, metal basis, ABCR) were used.

Preparation of Lead(II) Halide ILs. The list of chemicals and their purity can be found in the Supporting Information. All chemicals were used as received without further purification. ILs and lead halide salts were stored in an N2-filled glovebox to avoid moisture, as the neat ionic liquids are hygroscopic. In a typical synthetic procedure of nTHTDP-PbX2, PbCl2 or PbBr2 powders were mixed with a required volume of THTDPCI or THTDPBr, correspondingly. To aid the dissolution of lead halides, suspensions were stirred with a magnetic stirrer and heated to around 100–120 °C on a hot plate. In such conditions, samples with \(X_{\text{PbX2}} \leq 0.25\) can be

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prepared in about 30 min, while samples with $X_{PbX_2} \geq 0.33$ require overnight stirring. For the sTBPX-PbX$_2$, wherein TBPX are solids, alkyl phosphonium halides and lead halides were mixed in the desired ratio and heated to about 100–120 °C on a hot plate. After several minutes, the TBPX salts melted and PbX$_2$ dissolution continued upon stirring. The neat ILs were stored in the glovebox for further analysis and experiments.

PL and PLE steady-state spectra were recorded either with FluoroMax4-Plus-P (Horiba Jobin Yvon) or Fluorolog 3 (Horiba Jobin Yvon), equipped with 150 and 500 W xenon lamps, respectively.

UV–vis absorbance spectra were collected on a Jasco V670 spectrophotometer equipped with a deuterium (D2) lamp (190–350 nm) for use in UV, a halogen lamp (330–2700 nm) for use in UV/NIR, and an integrating sphere (ILN-725) with a working wavelength range of 220–2200 nm. Ionic liquids were drop cast between two quartz slides.

PL decay traces were recorded with a FluTime 300 spectrometer from PicoQuant Gmbh equipped with a TimeHarp 260 PICO counting unit. As an excitation source for time-resolved PL traces, a frequency-tripled Nd:YAG laser (wavelength of 355 nm, a power density of about 10 mW cm$^{-2}$, pulse duration 10 ps) was used. A scattering light from excitation was suppressed on the entrance of the emission monochromator by a 400 nm long-pass filter. The samples were measured in a quartz cuvette with a 10 mm path length.

Absolute PLQYs were measured using a Quantaurus-QY spectrometer from Hamamatsu.

**Fast Neutron Imaging at NECTAR.** Fast neutron imaging was conducted at the FRM-II reactor (Garching, Germany) on the NECTAR (neutron computed tomography and radiography) beamline. A schematic of the instrument and a fast neutron imaging test, the scintillator enclosure was sealed with a thin aluminum plate, to which a cuvette sample holder was attached with aluminum tape. The scintillator samples were placed into cuvettes and sealed to the enclosure with aluminum tape. Light yield measurements were taken of an empty sample stage, while thick metal blocks (either 1.5-cm-thick W, 5-cm-thick Fe, or 2.5-cm-thick Pb) were placed in the beam path to provide a sharp edge for spatial resolution measurements. Note that a pixel represents approximately 0.1 mm with the current camera focus, determined by the width of the 10 mm cuvettes at 103 pixels (0.097 mm/pixel). Images were processed using the Fiji distribution of ImageJ.

**Fast Neutron Imaging at NEUTRA (PSI).** A follow-up fast neutron imaging experiment to test the light yield of the highest concentration IL sample was conducted at the NEUTRA beamline of the Swiss spallation neutron source at the Paul Scherrer Institute (Villigen, Switzerland). Briefly, the cuvettes with the ionic liquids and the reference ZnS:Cu with a thin aluminum tape. Light yield measurements were taken of an empty sample stage, while thick metal blocks (either 1.5-cm-thick W, 5-cm-thick Fe, or 2.5-cm-thick Pb) were placed in the beam path to provide a sharp edge for spatial resolution measurements. Note that a pixel represents approximately 0.1 mm with the current camera focus, determined by the width of the 10 mm cuvettes at 103 pixels (0.097 mm/pixel). Images were processed using the Fiji distribution of ImageJ.

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Notes

The authors declare no competing financial interest.

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**REFERENCES**

(1) Dangendorf, V.; Zboray, R. Potential and Status in Imaging with Fast Neutrons. *Neutron News* **2015**, 26 (2), 27–30.

(2) Makowska, M. G.; Walfort, B.; Zboray, A.; Grunzweig, C.; Bucherl, T. Performance of the Commercial PP/ZnS:Cu and PP/ZnS:Ag Scintillation Screens for Fast Neutron Imaging. *J. Imaging* **2017**, 3 (4), 60.

(3) McCull, K. M.; Sakhatskyi, K.; Lehmann, E.; Walfort, B.; Losko, A. A.; Montanarella, F.; Bodnarchuk, M. I.; Cieslak, F.; Kozek, D.; Shynkarenko, Y.; Yakunin, S.; Kovalenko, M. V. Fast Neutron Imaging with Semiconductor Nanocrystal Scintillators. *ACS Nano* **2020**, 14 (11), 14686–14697.

(4) Kakavelakis, G.; Goda, M.; Sotiriou, A.; Kymakis, E.; Anthopoulos, T. D.; Petrides, K. Metal Halide Perovskites for High-Energy Detection. *Adv. Sci.* **2020**, 7, 2002098.

(5) Stoumpos, C. C.; Kanatzidis, M. G. The Renaissance of Halide Perovskites and Their Evolution as Emerging Semiconductors. *Acc. Chem. Res.* **2015**, 48 (10), 2791–2802.

(6) Sarapov, B.; Mitzi, D. B. Organic-Inorganic Perovskites: Structural Versatility for Functional Materials Design. *Chem. Rev.* **2016**, 116 (7), 4558–4596.

(7) Smith, M. D.; Connor, B. A.; Karunadasa, H. I. Tuning the Luminescence of Layered Halide Perovskites. *Chem. Rev.* **2019**, 119 (5), 3104–3139.

(8) Lin, H.; Zhou, C.; Tian, Y.; Siegrist, T.; Ma, B. Low-Dimensional Organometal Halide Perovskites. *ACS Energy Lett.* **2018**, 3 (1), 54–62.

(9) He, Q.; Zhou, C.; Xu, L.; Lee, S. J.; Lin, X. S.; Neu, J.; Worke, M.; Chaaban, M.; Ma, B. W. Highly Stable Organic Antimony Halide Crystals for X-ray Scintillation. *ACS Mater. Lett.* **2020**, 2 (6), 633–638.

(10) Morad, V.; Shynkarenko, Y.; Yakunin, S.; Brumberg, A.; Schaller, R. D.; Kovalenko, M. V. Disphenoidal Zero-Dimensional Lead, Tin, and Germanium Halides: Highly Emissive Single and Triplet Self-Trapped Excitons and X-ray Scintillation. *J. Am. Chem. Soc.* **2019**, 141 (25), 9764–9768.

(11) Xu, L.-J.; Lin, X.; He, Q.; Worke, M.; Ma, B. Highly Efficient Eco-Friendly X-ray Scintillators Based on an Organic Manganese Halide. *Nat. Commun.* **2020**, 11 (1), 4329.

(12) Lian, L.; Zhong, M. Y.; Zhang, W. Z.; Yin, L. X.; Du, X. Y.; Zhang, P.; Zhang, X. W.; Gao, J. B.; Zhang, D. L.; Gao, L.; Niu, G. D.; Song, H. S.; Chen, B. R.; Lan, X. Z.; Tang, J.; Zhang, J. B. Efficient and Reabsorption-Free Radioluminescence in Cs₂Cu₂I₅ Nanocrystals with Self-Trapped Excitons. *Adv. Sci.* **2020**, 7 (11).

(13) Zhou, C. K.; Lin, H. R.; Neu, J.; Zhou, Y.; Chaaban, M.; Lee, S.; Worke, M.; Chen, B. H.; Clark, R.; Cheng, W. H.; Guan, J. J.; Djuricovich, P.; Zhang, D. Z.; Lu, X. J.; Bullock, J.; Pak, C.; Shatruk, M.; Du, M. H.; Siegert, T.; Ma, B. W. Green Emitting Single-Crystalline Bulk Assembly of Metal Halide Clusters with Near-Unity Photoluminescence Quantum Efficiency. *ACS Energy Lett.* **2019**, 4 (7), 1579–1583.

(14) Niki, M.; Mihokova, E.; Nitsch, K. Photoluminescence & Decay Kinetics of Cs₅PbBr₃ Single Crystals. *Solid State Commun.* **1992**, 82 (12), 1089–1092.

(15) Niki, M.; Mihokova, E.; Nitsch, K.; Somma, F.; Giampaolo, C.; Pazzi, G. P.; Fabeni, P.; Zazubovich, S. Photoluminescence of Cs₅PbBr₃ crystals and thin films. *Chem. Phys. Lett.* **1999**, 306 (5), 280–284.

(16) Han, D.; Shi, H. L.; Ming, W. M.; Zhou, C. K.; Ma, B. W.; Sarapov, B.; Ma, Y. Z.; Chen, S. Y.; Du, M. H. Unraveling Luminescence Mechanisms in Zero-Dimensional Halide Perovskites. *J. Mater. Chem. C* **2018**, 6 (24), 6398–6405.

(17) Bohun, A.; Dolejší, J.; Barta, Č. The Absorption and Luminescence of (PbCl₆)₃⁻ and (PbBr₆)³⁻ Complexes. *Czech. J. Phys.* **2017**, 70 (7), 803–807.

(18) Nikol, H.; Becht, A.; Vogler, A. Photoluminescence of Germanium(II), Tin(II), and Lead(II) Chloride Complexes in Solution. *Inorg. Chem.** 1992**, 31 (15), 3277–3279.

(19) Akkerman, Q. A.; Park, S.; Radicchi, E.; Nunzi, F.; Mosconi, E.; De Angelis, F.; Brescia, R.; Rastogi, P.; Prato, M.; Manna, L. Nearly Monodisperse Insulator Cs₂PbX₆ (X = Cl, Br, I) Nanocrystals, Their Mixed Halide Compositions, and Their Transformation into CsPbX₆ Nanocrystals. *Nano Lett.* **2017**, 17 (3), 1924–1930.

(20) Tan, L.; Wang, W.; Li, Q.; Luo, Z.; Zou, C.; Tang, M.; Zhang, L.; He, J.; Quan, Z. Colloidal Syntheses of Zero-Dimensional Cs₂SnX₆ (X = Br, I) Nanocrystals with High Emission Efficiencies. *Chem. Commun.* **2020**, 56 (3), 387–390.

(21) Cheng, P.; Sun, L.; Feng, L.; Yang, S.; Yang, Y.; Zheng, D.; Zhao, Y.; Sang, Y.; Zhang, R.; Wei, D.; Deng, W.; Han, K. Colloidal Synthesis and Optical Properties of All-Inorganic Low-Dimensional Cesium Copper Halide Nanocrystals. *Angew. Chem., Int. Ed.* **2019**, 58 (45), 16087–16091.

(22) Wang, A.; Yan, X.; Zhang, M.; Sun, S.; Yang, M.; Shen, W.; Pan, X.; Wang, P.; Deng, Z. Controlled Synthesis of Lead-Free and Stable Perovskite Derivative Cs₂SnI₆ Nanocrystals via a Facile Hot-Injection Process. *Chem. Mater.* **2016**, 28 (22), 8132–8140.

(23) MacFarlane, D. B. K.; Pringle, J. M. An Introduction to Ionic Liquids. In *Fundamentals of Ionic Liquids*; Wiley-VCH Verlag GmbH & Co. KGaA: Weinheim, Germany, 2017; pp 1–25.

(24) Estager, J.; Holbrey, J. D.; Swadzba-Kwasny, M. Halometallate Ionic Liquids - Revisited. *Chem. Soc. Rev.* **2014**, 43 (3), 847–886.

(25) Kore, R.; Berton, P.; Kelley, S. P.; Aduri, P.; Katti, S. S.; Rogers, R. D. Group IIA Halometallate Ionic Liquids: Speciation and Applications in Catalysis. *ACS Catal.* **2017**, 7 (10), 7014–7028.

(26) Yoshida, Y.; Saito, G. Design of Functional Ionic Liquids Using Magneto- and Luminescent-Active Anions. *Phys. Chem. Chem. Phys.* **2010**, 12 (8), 1675–1684.

(27) Alvarez-Vicente, J.; Jandil, S.; Banerjee, D.; Gunaratne, H. Q. N.; Gray, S.; Felton, S.; Strinivasan, G.; Kaczmarek, A. M.; Van Deun, R.; Nockemann, P. Easily Accessible Rare-Earth-Containing Phosphonium Room-Temperature Ionic Liquids: EXAFS, Luminescence, and Magnetic Properties. *J. Phys. Chem. B* **2016**, 120 (23), 5301–5311.
(28) Coleman, F.; Feng, G.; Murphy, R. W.; Nockemann, P.; Seddon, K. R.; Swadzba-Kwasny, M. Lead(II) Chloride Ionic Liquids and Organic/Inorganic Hybrid Materials - a Study of Chloroplumbate(II) Speciation. *Dalton T.* 2013, 42 (14), S025–S035.

(29) Zhou, C. C.; Lin, H. R.; Worku, M.; Neu, J.; Zhou, Y.; Tian, Y.; Lee, S.; Djurovich, P.; Siegrist, T.; Ma, B. Blue Emitting Single Crystalline Assembly of Metal Halide Clusters. *J. Am. Chem. Soc.* 2018, 140 (41), 13181–13184.

(30) Oldenburg, K.; Vogler, A. Electronic-Spectra and Photochemistry of Tin(II), Lead(II), Antimony(III), and Bismuth(III) Bromide Complexes in Solution. *Z. Naturforsch., B: J. Chem. Sci.* 1993, 48 (11), 1519–1523.

(31) Chin, S.-H.; Choi, J. W.; Hu, Z.; Mardegan, L.; Sessolo, M.; Bolink, H. J. Tunable Luminescent Lead Bromide Complexes. *J. Mater. Chem. C* 2020, 8 (45), 15996–16000.

(32) Byrne, R. H.; Yao, W. S.; Luo, Y. X.; Millero, F. J. Complexation of Pb(II) by Chloride Ions in Aqueous Solutions. *Aquat. Geochem.* 2010, 16 (3), 325–335.

(33) Stampelescoskie, K. G.; Manser, J. S.; Kamat, P. V. Dual Nature of the Excited State in Organic-Inorganic Lead Halide Perovskites. *Energy Environ. Sci.* 2015, 8 (1), 208–215.

(34) Morad, V.; Yakunin, S.; Kovalenko, M. V. Supramolecular Approach for Fine-Tuning of the Bright Luminescence from Zero-Dimensional Antimony(III) Halides. *ACS Mater. Lett.* 2020, 2 (7), 845–852.

(35) Zboray, R.; Adams, R.; Morgano, M.; Kis, Z. Qualification and Development of Fast Neutron Imaging Scintillator Screens. *Nucl. Instrum. Methods Phys. Res., Sect. A* 2019, 930, 142–150.

(36) Buffler, A.; Tickner, J. Detecting Contraband Using Neutrons: Challenges and Future Directions. *Radiat. Meas.* 2010, 45 (10), 1186–1192.

(37) Kardjilov, N.; Manke, I.; Woracek, R.; Hilger, A.; Banhart, J. Advances in Neutron Imaging. *Mater. Today* 2018, 21 (6), 652–672.

(38) Bücherl, T.; Sollradi, S. NECTAR: Radiography and Tomography Station using Fission Neutrons. *JLSRF* 2015, 1, A19.

(39) Lehmann, E. H.; Vontobel, P.; Wiesel, L. Properties of the Radiography Facility NEUTRA at SINQ and its Potential for Use as European Reference Facility. *Nondestruct. Test. Nondestr. Test. Eval.* 2001, 16 (2–6), 191–202.

(40) McCall, K. M.; Sakhatskyi, K.; Lehmann, E.; Walfort, B.; Losko, A. S.; Montanarella, F.; Bodnarchuk, M. I.; Krieg, F.; Kelestemur, Y.; Mannes, D.; Shynkarenko, Y.; Yakunin, S.; Kovalenko, M. V. Fast Neutron Imaging with Semiconductor Nanocrystal Scintillators. *ACS Nano* 2020, 14, 14686.

(41) Makowska, M. G.; Walfort, B.; Zeller, A.; Grünzweig, C.; Bücherl, T. Performance of the Commercial PP/ZnS: Cu and PP/ZnS: Ag Scintillation Screens for Fast Neutron Imaging. *J. Imaging* 2017, 3 (4), 60.

(42) Makii, H.; Nishio, K.; Hirose, K.; Orlandi, R.; Léguillon, R.; Ogawa, T.; Soldner, T.; Hambisch, F.-J.; Astier, A.; Pollitt, A.; Petrache, C.; Tsekhanovich, I.; Mathieu, L.; Aiche, M.; Frost, R.; Czajkowski, S.; Guo, S.; Köster, U. Measurement of High-Energy Prompt Gamma-Rays from Neutron Induced Fission of U-235. *EPJ Web Conf.* 2017, 146, 04036.

(43) JEF Report 14: Table of Simple Integral Neutron Cross Section Data from JEF-2.2, ENDF/B-VI, JENDL-3.2, BROND-2 and CENDL-2. [http://www.oecd-nea.org/dbdata/nds_jefreports/](http://www.oecd-nea.org/dbdata/nds_jefreports/); accessed June 3, 2020.

(44) Andor’s iKon-L High Dynamic Range CCD Camera. [https://andor.oxinst.com/products/ikon-xl-and-ikon-large-ccd-series/ikon-l-936](https://andor.oxinst.com/products/ikon-xl-and-ikon-large-ccd-series/ikon-l-936); accessed June 30, 2020.

(45) Schindelin, J.; Arganda-Carreras, I.; Frise, E.; Kaynig, V.; Longair, M.; Pietzsch, T.; Preibisch, S.; Rueden, C.; Saalfeld, S.; Schmid, B.; Tinevez, J.-Y.; White, D. J.; Hartenstein, V.; Eliceiri, K.; Tomancak, P.; Cardona, A. Fiji: an Open-Source Platform for Biological-Image Analysis. *Nat. Methods* 2012, 9 (7), 676–682.

(46) Schneider, C. A.; Rasband, W. S.; Eliceiri, K. W. NIH Image to ImageJ: 25 Years of Image Analysis. *Nat. Methods* 2012, 9 (7), 671–675.