Effect of Anionic Group \([\text{SiO}_4]^{4-}/[\text{PO}_4]^{3-}\) on the Luminescence Properties of \(\text{Dy}^{3+}\)-Doped Tungstate Structural Compounds

Ning Liu,\(^1,2\) Lefu Mei,\(^2,\ast\) Jianxiong Bin,\(^2\) Zhijian Peng\(^{1,\ast}\)

\(^1\)School of Science, China University of Geosciences, Beijing 100083, P.R. China.

\(^2\)Beijing Key Laboratory of Materials Utilization of Nonmetallic Minerals and Solid Wastes, National Laboratory of Mineral Materials, School of Materials Sciences and Technology, China University of Geosciences, Beijing 100083, P.R. China

\(\ast\)Corresponding authors

Email: \texttt{mlf@cugb.edu.cn} (Lefu Mei)

Email: \texttt{pengzhijian@cugb.edu.cn} (Zhijian Peng)
Abstract: The novel scheelite structures of Li$_2$Ca(WO$_4$)$_2$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$), and LiCa$_2$(WO$_4$)(PO$_4$) fluorescent materials were successfully prepared using a high-temperature solid-phase method. The compounds were characterized by XRD, EDS, and XRF. The test results showed that the substitution of [WO$_4$]$^{2-}$ by [SiO$_4$]$^{4-}$/[PO$_4$]$^{3-}$ tetrahedron in tungstate had no significant influence on the crystal structure of the Li$_2$Ca(WO$_4$)$_2$. When Dy$^{3+}$ ions were introduced as an activator at an optimum doping concentration of 0.08% mol, all of the as-prepared phosphors generated yellow light emissions, and the emission peak was located close to 576 nm. Replacing [WO$_4$]$^{2-}$ with [SiO$_4$]$^{4-}$/[PO$_4$]$^{3-}$ tetrahedron significantly increased the luminescence of the Li$_2$Ca(WO$_4$)$_2$ phosphors. In addition, the LiCa$_2$(WO$_4$)(PO$_4$):zDy$^{3+}$ phosphors had the best luminescence properties, decay life (τ = 0.049 ms), and thermal stability (87.8%).

Keywords: phosphors; scheelite structure; luminescence property; anion substitution

1. Introduction

Inorganic luminescent materials have historically attracted considerable attention due to their characteristics such as long decay lifetimes [1], easy preparation, low toxicity, and high luminous efficiency [2-4]. Rare earth luminescent materials have considerable application potential because of their unique 4f orbit, which can be pumped to excited energy levels to emit colorful light under excitation [5, 6]. Among them, the emission of Dy$^{3+}$ is mainly from the yellow-orange region (570-600 nm), $^4$F$_{9/2}$$\rightarrow$$^6$H$_{13/2}$ transition [7], and $^4$F$_{9/2}$$\rightarrow$$^6$H$_{15/2}$ transition at the blue region (470-500 nm) [8]. Dy$^{3+}$ can also emit white light at a suitable yellow-blue intensity ratio [9]. The
excitation spectrum of Dy$^{3+}$ consists only of narrow $f$-$f$ transition lines with low oscillation intensity due to the characteristics of the 300-500 nm parity selection rule (Dy$^{3+}$ charge transfer absorption bands or 4f$^8$→4f$^7$5d bands below 200 nm). In the matrix sensitization of Dy$^{3+}$ ions, efficient energy transfer from the matrix to the Dy$^{3+}$ ions is one of the most important methods of achieving efficient emission of Dy$^{3+}$ ions [10].

The luminescent properties of phosphors are not only affected by rare earth ions, but are also closely related to the structure and composition of the material hosts [11, 12]. Tungstate materials are considered to be effective self-activated fluorescent materials with high-absorption cross-sections [13, 14], broadband emissions [15] and high quantum efficiency [16, 17]. Self-activating compounds can also act as matrix sensitizers to address weak $f$-$f$ transitions in trivalent rare earth ions. The matrix material’s lattice not only affects the dopant’s optical transition, but also reduces its excitation energy [18]. Among various matrix materials, phosphate has a wide band gap and high efficiency, with high thermal stability and chemical durability [19, 20]. Silicate has been widely studied as a satisfactory matrix material for fluorescent materials due to its excellent optical properties, chemical stability, and structural diversity [21, 22]. Based on these advantages, [SiO$_4$]$^{4-}$ or [PO$_4$]$^{3-}$ ion substitution of [WO$_4$]$^{2-}$ ion to synthesize a novel scheelite structural matrix in tungstate is considered a viable and suitable luminescent material carrier [23]. Combining the stable physical and chemical properties of the matrix materials, Li$_2$Ca(WO$_4$)$_2$:xDy$^{3+}$ and Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):yDy$^{3+}$, LiCa$_2$(WO$_4$)(PO$_4$):zDy$^{3+}$ phosphors with low energy consumption and high thermal
stability were synthesized using a high-temperature solid-phase method, and the luminescence behavior of Dy\(^{3+}\) in the three hosts was studied. To verify the effect of [SiO\(_4\)]\(^4-\) or [PO\(_4\)]\(^3-\) substitution on the tungstate matrix fluorescent material’s properties and elucidate the effect of anion substitution on the crystal structure, a series of comparative experiments were conducted to study the various properties after the replacement of the samples’ crystal phase and anions. The luminescence properties of the doped ion samples are discussed, including excitation spectrum, emission spectrum, decay time, chromaticity diagram, and luminescence mechanism of the doping ions in three modified matrices. In addition, the thermal stability was tested to explore potential practical applications.

2. Experimental section

2.1. Materials and preparation

All of the as-prepared solid-solution samples were synthesized using a conventional high-temperature solid-phase reaction under an air atmosphere. The initial raw materials were WO\(_3\) (analytical reagent [AR]), SiO\(_2\) (AR), Dy\(_2\)O\(_3\) (99.99%), CaCO\(_3\) (AR), Li\(_2\)CO\(_3\) (AR), and (NH\(_4\))\(_2\)HPO\(_4\) (AR) (excessive H\(_3\)BO\(_3\) was added as flux) of analytic grade purity from Beijing Chemical Co. China. All of the initial reagents were weighed according to the stoichiometric ratios and mixed by grinding in an agate mortar for more than 10 minutes and pre-sintered in a muffle furnace at 550 °C for 2 hours. The sample then was cooled to room temperature and thoroughly ground again for 5 minutes. Next, different anion-regulated compounds were sintered under varying conditions. The Li\(_2\)Ca(WO\(_4\))\(_{2-x}\)Dy\(^{3+}\) sample was prepared via calcination at 1000 °C
for 3 hours in a high-temperature tube furnace at a heating rate of 5 °C/min. The Li₂Ca₂(WO₄)(SiO₄):yDy³⁺ phosphor was synthesized in a high-temperature tube furnace. It was then calcined at the same heating rate to 850 °C for 3 hours, while the LiCa₂(WO₄)(PO₄):zDy³⁺ phosphor was calcined at 1400 °C for 3 hours. The samples were then naturally cooled to room temperature and thoroughly ground in an agate mortar for further measurements.

2.2. Characterization techniques

The structural properties of the as-prepared phosphors were determined using a Bruker D8 (Germany) powder X-ray diffractometer (XRD) with Cu Kα radiation (λ = 0.15406 nm) working at 40 kV and 40 mA in a range from 10° to 70°. The step scanning rate (2θ ranging from 5° to 130°) was 2.5 s/step with a step size of 0.02°. The excitation and emission spectra of the as-prepared samples were measured using a spectrophotometer (F-4600, HITACHI, Japan) equipped with a 150 W Xe lamp operating at 400 V. High-resolution transmission electron microscopy (TEM) measurements were carried out using a Tecnai G2 transmission electron microscope (TMP) operating at 200 kV. The samples’ decay curves were recorded by a spectrofluorometer (HORIBA, Jobin Yvon FL3-21). The temperature-dependent PL spectra were determined on a Hitachi F-4600 spectrophotometer equipped with a self-made heating set-up and a self-controlled electric incinerator.

3. Results and discussion

3.1. Crystal structure
The crystalline structures of the synthesized samples are shown in Figure 1. The Li$_2$Ca(WO$_4$)$_2$:xDy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$)$_y$:yDy$^{3+}$, and LiCa$_2$(WO$_4$)(PO$_4$)$_z$:zDy$^{3+}$ phases were all pure phases. The samples’ XRD patterns corresponded to calcium tungstate (CaWO$_4$, JCPDS no. 41-1431) and the second phases were not detected, indicating that the [SiO$_4$]$^{4-}$/[PO$_4$]$^{3-}$ successfully replaced [WO$_4$]$^{2-}$ in the tungstate and was incorporated into the matrix structure. The introduction of Dy$^{3+}$ ions caused no significant changes in the three matrix lattices.

3.2. Morphology and elemental analysis

Energy dispersive spectroscopy (EDS) and X-ray fluorescence spectroscopy (XRF) were carried out to assess the chemical composition of the compounds prepared in this study. Figure 2(a) shows a scanning electron microscope (SEM) image of the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) sample and Figure 2(b) shows an EDS elemental analysis of the surface points in a rectangular region. The sample’s particles were not uniform and relatively large due to agglomeration during heating. In addition, the mapping result (Figure 2(c)) demonstrated the presence of Si$^{4+}$ ions in the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$), and the distribution of the elements including Si was very uniform, indicating that [SiO$_4$]$^{4-}$ successfully replaced the [WO$_4$]$^{2-}$ position. The samples prepared in the experiment were the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) phosphor we designed. The same examination were also carried out in the LiCa$_2$(WO$_4$)(PO$_4$) compounds. Figure 2(d) shows an agglomeration scanning electron microscope (SEM) image of the Li$_2$Ca$_2$(WO$_4$)(PO$_4$) sample and Figure 2(e) shows an EDS elemental analysis of the surface points in a rectangular region. As demonstrated in Figure 2(f), the presence of elements such as Ca, O, W, P,
and Au was detected by mapping in the LiCa$_2$(WO$_4$)(PO$_4$) compounds, and all were also uniform. The results showed that no additional impurity elements were present except for Au in the SEM sample holder. The elements were evenly distributed, indicating that [PO$_4$]$^{4-}$ also successfully replaced the [WO$_4$]$^{2-}$ position, and the sample prepared in the experiment was indeed the LiCa$_2$(WO$_4$)(PO$_4$) compound. Tables 1 and Tables 2 shows the chemical elemental analysis results of the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) and LiCa$_2$(WO$_4$)(PO$_4$) samples, respectively. The data indicated that the content ratios of Ca, O, W, and Si in the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) samples were 16.2 mass%, 38.7 mass%, 36.6 mass%, and 6.07 mass%, respectively, while the content ratios of Ca, O, W, and P in the LiCa$_2$(WO$_4$)(PO$_4$) samples were 16.0 mass%, 35.8 mass%, 34.3 mass%, and 8.58 mass%, respectively. The amount of each element was very close to the theoretical content. The results obtained by combining the EDS and XRF spectra tests clearly demonstrated that the prepared sample had a pure phase scheelite structural matrix.

3.3. Energy transfer processes involved in Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):$_y$Dy$^{3+}$ and LiCa$_2$(WO$_4$)(PO$_4$):$_z$Dy$^{3+}$ phosphors

As an important activator, Dy$^{3+}$ ions have been studied in detail in many matrices [24, 25]. For scheelite structural matrices, Dy$^{3+}$ ions can also be incorporated into the crystal lattice as an effective activator. Figures 3(a) and 3(b) show the curves of the characteristic yellow emission peak (576 nm) and PL emission spectrum of the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):$_y$Dy$^{3+}$ and LiCa$_2$(WO$_4$)(PO$_4$):$_z$Dy$^{3+}$ phosphors excited by 353 nm light with a Dy$^{3+}$ ion doping concentration. Since the Dy$^{3+}$ ion 4f-4f allowed transitions, the sample’s PLE spectrum contained many peaks in the 200-450 nm region, including
broadband emission peaks appearing at approximately 228, 327, 353, 366, and 388 nm. At 353 nm excitation, the emission spectra of the Li$_2$Ca$_2$(WO$_4$)(SiO$_4$)$_y$Dy$^{3+}$ and LiCa$_2$(WO$_4$)(PO$_4$)$_z$Dy$^{3+}$ consisted of characteristic electronic transitions of Dy$^{3+}$ ions corresponding to the blue emission of the magnetic dipole $^4F_{9/2}$$\rightarrow$$^6H_{15/2}$ (478 and 488 nm) transitions. The yellow emission demonstrated the electric dipole $^4F_{9/2}$$\rightarrow$$^6F_{13/2}$ (576 nm) transitions. The energy difference between $^4F_{9/2}$ and $^6F_{3/2}$ is close to the energy difference between $^6H_{15/2}$ and $^6H_{9/2}$, which usually leads to cross-relaxation of the resonance energy transfer process: Dy$^{3+}$ ($^4F_{9/2} + $Dy$^{3+}$ $^6H_{15/2}$$\rightarrow$Dy$^{3+}$ ($^6F_{3/2}$) + Dy$^{3+}$ ($^6H_{9/2}$). As shown in Figures 3(a) and 3(b), the emission intensity of the $^4F_{9/2}$$\rightarrow$$^6F_{13/2}$ transition first increased as the Dy$^{3+}$ concentration increased, and then began to decrease at a concentration of 0.08% mol because of the effective concentration at 353 nm excitation. Figure 4 shows the contrast of the emission spectra of the Li$_2$Ca(WO$_4$)$_2$:xDy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$)$_y$Dy$^{3+}$, and LiCa$_2$(WO$_4$)(PO$_4$)$_z$Dy$^{3+}$ samples excited by 353 nm light. As shown in the inset, the LiCa$_2$(WO$_4$)(PO$_4$)$_z$Dy$^{3+}$ had the best luminescence intensity, indicating that [PO$_4$]$^{3-}$ ion replacement [WO$_4$]$^{2-}$ can significantly improve the performance of tungstate matrix fluorescent materials.

3.4 Effect of [SiO$_4$]$^{4-}$ and [PO$_4$]$^{3-}$substituting [WO$_4$]$^{2-}$ ions on the thermal stability of phosphors

Thermal stability is a key parameter to consider in practical high-power solid-state lighting applications [26-28]. The thermal stability of the Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$)$_y$:0.08Dy$^{3+}$, and LiCa$_2$(WO$_4$)(PO$_4$)$_z$:0.08Dy$^{3+}$ samples was evaluated according to their temperature dependence. The emission spectrum obtained using 353
nm wavelength light was used to produce excitation light in a temperature range of 30 °C to 300 °C. Figure 5 clearly shows the dependence of the luminescence intensity of the Dy³⁺ (576 nm) on different temperatures. As the temperature increased, the sample’s luminescence intensity gradually decreased and thermal quenching occurred. The changes in the three phosphors’ luminescence intensity differed from each other, indicating that the substitution of anions can significantly change the thermal stability of tungstate fluorescent materials[29]. When the temperature increased to 150 °C, the thermal stability of the Li₂Ca(WO₄)₂:0.08Dy³⁺, Li₂Ca₂(WO₄)(SiO₄):0.08Dy³⁺, and LiCa₂(WO₄)(PO₄):0.08Dy³⁺ samples was 79.85%, 75.85%, and 87.8%, respectively. The luminescence intensity of the LiCa₂(WO₄)(PO₄):0.08Dy³⁺ sample was less affected by temperature changes than the other two samples, indicating that the LiCa₂(WO₄)(PO₄):0.08Dy³⁺ had the best thermal stability among the three kinds of fluorescent materials, and all three samples had high thermal stability.

3.5 Fluorescence lifetime analysis

We further investigated the effect of anion substitution on the luminescence properties and fluorescence lifetimes of the scheelite structural compounds. According to the experimental data, the attenuation curves of the three compounds exhibited a second-order exponential decay that can be fit with the following function[30]:

\[ I(t) = I_0 + A_1 \exp(-t/\tau_1) + A_2 \exp(-t/\tau_2) \]
where $I(t)$ is the luminescence intensity at time $t$, $A_1$ and $A_2$ are the fitting parameters, $t$ is the time, and $\tau_1$ and $\tau_2$ are the lifetimes of the exponential components, respectively. The average lifetime $\tau^*$ is calculated using the following equation:

$$\tau^* = \frac{A_1\tau_1^2 + A_2\tau_2^2}{A_1\tau_1 + A_2\tau_2}$$

The $\text{Li}_2\text{Ca}\text{(WO}_4\text{)}_2\cdot0.08\text{Dy}^{3+}$, $\text{Li}_2\text{Ca}_2\text{(WO}_4\text{)(SiO}_4\text{)}\cdot0.08\text{Dy}^{3+}$, and $\text{LiCa}_2\text{(WO}_4\text{)(PO}_4\text{)}\cdot0.08\text{Dy}^{3+}$ samples ($\lambda_{\text{ex}} = 353$ nm and $\lambda_{\text{em}} = 576$ nm) were tested for fluorescence decay life. Figure 6 shows that the average decay time ($\tau^*$) of the $\text{Li}_2\text{Ca}(\text{WO}_4)_2\cdot0.08\text{Dy}^{3+}$ phosphor emission was 0.032 ms. With $[\text{SiO}_4]^{4-}$ or $[\text{PO}_4]^{3-}$ instead of $[\text{WO}_4]^{2-}$, the fluorescence lifetime $\tau$ values of the $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4)\cdot0.08\text{Dy}^{3+}$ and $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)\cdot0.08\text{Dy}^{3+}$ were 0.024 ms and 0.049 ms, respectively. The $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)\cdot0.08\text{Dy}^{3+}$ sample had the best fluorescence lifetime, which matched the previously described variations in the PL spectrum and thermal stability[31].

3.6. CIE spectrum coordination

Figure 7 shows CIE chromaticity diagrams of the $\text{Li}_2\text{Ca}(\text{WO}_4)_2\cdot0.08\text{Dy}^{3+}$, $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4)\cdot0.08\text{Dy}^{3+}$, and $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)\cdot0.08\text{Dy}^{3+}$ samples and their corresponding photographs. The calculations demonstrated that the CIE chromaticity coordinates of the $\text{Li}_2\text{Ca}(\text{WO}_4)_2\cdot0.08\text{Dy}^{3+}$, $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4)\cdot0.08\text{Dy}^{3+}$, and $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)\cdot0.08\text{Dy}^{3+}$ phosphors were 0.4053 and 0.4523, 0.3892 and 0.4215, and 0.4124 and 0.4603, respectively, corresponding to the yellow light area. In these systems, $[\text{SiO}_4]^{4-}$ or $[\text{PO}_4]^{3-}$ substituted $[\text{WO}_4]^{2-}$, and the CIE chromaticity coordinates changed slightly, but the change was not large because the luminescent color of the
fluorescent material was mainly determined by the type of doped rare earth ions[32]. Therefore, the yellow light-emitting phosphor that was effectively excited by the near-ultraviolet light chip was obtained by doping the aforementioned matrix material with Dy$^{3+}$ ions, and the three kinds of fluorescent materials prepared have potential application materials as yellow phosphors for white LEDs.

4. Conclusion

The pure phases of the Dy$^{3+}$-doped scheelite structure Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):0.08Dy$^{3+}$, and LiCa$_2$(WO$_4$)(PO$_4$):0.08Dy$^{3+}$ phosphors were successfully prepared using a high-temperature solid-phase reaction. The prepared phosphors had a wide excitation band in the 320-430 nm band, which could be matched with commercial n-UV LED chips. These phosphors mainly emit yellow light peaks at 576 nm, and the optimum doping concentration was determined to be 0.08 % mol. When [SiO$_4$]$^{4-}$/[PO$_4$]$^{3-}$ substituted [WO$_4$]$^{2-}$, the luminescent properties, thermal stability, and lifetime decay curves of the phosphors were modified. The results showed that the as-prepared fluorescent materials had excellent thermal stability and can be used as potential w-LED luminescent materials.

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Table captions

Table 1 Chemical analysis of fractions of Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) (O, P, Ca, W).

Table 2 Chemical analysis of fractions of LiCa$_2$(WO$_4$)(PO$_4$) (O, Si, Ca, W).

### Table 1 Chemical analysis of fractions of Li$_2$Ca$_2$(WO$_4$)(SiO$_4$) (O, P, Ca, W).

| Chemical composition | Weight (%) |
|----------------------|------------|
| O                    | 29.79      |
| P                    | 7.21       |
| Ca                   | 18.62      |
| W                    | 42.77      |

### Table 2 Chemical analysis of fractions of LiCa$_2$(WO$_4$)(PO$_4$) (O, Si, Ca, W)

| Chemical composition | Weight (%) |
|----------------------|------------|
| O                    | 40.29      |
| Si                   | 6.31       |
| Ca                   | 16.88      |
| W                    | 38.05      |
Figure captions

Figure. 1. XRD pattern of Li$_2$Ca(WO$_4$)$_2$:xDy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):yDy$^{3+}$, LiCa$_2$(WO$_4$)(PO$_4$):zDy$^{3+}$ samples with the PDF#41-1431 (CaWO$_4$) as a reference.

Figure. 2. (a) EDS spectrum, (b) SEM image, (c) elemental mappings of the Li$_2$Ca$_2$WO$_4$SiO$_4$ phosphor; (d) EDS spectrum, (e) SEM image, (f) elemental mappings of the Li$_2$Ca$_2$WO$_4$SiO$_4$ phosphor.

Figure. 3. Excitation and emission spectra of spectra of (a) Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):yDy$^{3+}$, (b) LiCa$_2$(WO$_4$)(PO$_4$):zDy$^{3+}$ samples, (c) and (d) inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.

Figure. 4. PL spectra of (a) Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, (b) Li$_2$Ca$_2$(WO$_4$)(SiO$_4$): 0.08Dy$^{3+}$ and (c) LiCa$_2$(WO$_4$)(PO$_4$): 0.08Dy$^{3+}$ samples and (b) Inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.

Figure. 5. The PL spectra ($\lambda_{ex} = 353$ nm) of (a) Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, (b) Li$_2$Ca$_2$(WO$_4$)(SiO$_4$): 0.08Dy$^{3+}$ and (c) LiCa$_2$(WO$_4$)(PO$_4$): 0.08Dy$^{3+}$ samples under different temperatures in the range of 30–300 °C, every inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.

Figure. 6. Decay curves and lifetimes ($\tau$) of (a) Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, (b)
Li$_2$Ca$_2$(WO$_4$)(SiO$_4$): 0.08Dy$^{3+}$ and (c) LiCa$_2$(WO$_4$)(PO$_4$): 0.08Dy$^{3+}$ samples
($\lambda_{em}=576\text{nm}$) after the pulsed excitation ($\lambda_{ex} = 353\text{ nm}$) at RT.

Figure. 7. CIE color coordinates for the Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$ (orange star),
Li$_2$Ca$_2$(WO$_4$)(SiO$_4$): 0.08Dy$^{3+}$ (red star) and LiCa$_2$(WO$_4$)(PO$_4$): 0.08Dy$^{3+}$ (purple star) at RT. The results are calculated using the emission spectra under 353 nm excitation.
Figure 1. XRD pattern of $\text{Li}_2\text{Ca(WO}_4\text{)}_2:x\text{Dy}^{3+}$, $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4):y\text{Dy}^{3+}$, $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4):z\text{Dy}^{3+}$ samples with the PDF#41-1431 (CaWO$_4$) as a reference.
Figure 2. (a) EDS spectrum, (b) SEM image, (c) elemental mappings of the Li$_2$Ca$_2$WO$_4$SiO$_4$ phosphor; (d) EDS spectrum, (e) SEM image, (f) elemental mappings of the Li$_2$Ca$_2$WO$_4$SiO$_4$ phosphor.

Figure 3  Excitation and emission spectra of spectra of (a) Li$_2$Ca$_2$(WO$_4$)(SiO$_4$):yDy$^{3+}$, (b) LiCa$_2$(WO$_4$)(PO$_4$):zDy$^{3+}$ samples, (c) and (d) inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.
Figure 4. PL spectra of Li$_2$Ca(WO$_4$)$_2$:0.08Dy$^{3+}$, Li$_2$Ca$_2$(WO$_4$)(SiO$_4$): 0.08Dy$^{3+}$ and LiCa$_2$(WO$_4$)(PO$_4$): 0.08Dy$^{3+}$ samples and (b) Inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.
Figure. 5. The PL spectra ($\lambda_{ex} = 353$ nm) of (a) $\text{Li}_2\text{Ca(WO}_4\text{)}_2:0.08\text{Dy}^{3+}$, (b) $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4):0.08\text{Dy}^{3+}$ and (c) $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4):0.08\text{Dy}^{3+}$ samples under different temperatures in the range of 30–300 °C, every inset shows the corresponding dependence of PL intensity of Dy$^{3+}$ transition.
Figure. 6. Decay curves and lifetimes ($\tau$) of $\text{Li}_2\text{Ca}(\text{WO}_4)_2$:0.08$\text{Dy}^{3+}$, $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4)$: 0.08$\text{Dy}^{3+}$ and $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)$: 0.08$\text{Dy}^{3+}$ samples ($\lambda_{\text{em}}$=576 nm ) after the pulsed excitation ($\lambda_{\text{ex}}$ = 353 nm) at RT.

Figure. 7. CIE color coordinates for the $\text{Li}_2\text{Ca}(\text{WO}_4)_2$:0.08$\text{Dy}^{3+}$ (orange star) , $\text{Li}_2\text{Ca}_2(\text{WO}_4)(\text{SiO}_4)$: 0.08$\text{Dy}^{3+}$ (red star) and $\text{LiCa}_2(\text{WO}_4)(\text{PO}_4)$: 0.08$\text{Dy}^{3+}$ (purple star) at RT. The results are calculated using the emission spectra under 353 nm excitation.