Effect of doping and pressure on magnetism and lattice structure of iron-based superconductors

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Using first-principles calculations, we analyze structural and magnetic trends as a function of charge doping and pressure in BaFe$_2$As$_2$, and compare to experimentally established facts. We find that density-functional theory, while accurately reproducing the structural and magnetic ordering at ambient pressure, fails to reproduce some structural trends as pressure is increased. Most notably, the Fe-As bond length which is a gauge of the magnitude of the magnetic moment, $\mu$, is rigid in experiment but soft in calculation, indicating residual local Coulomb interactions. By calculating the magnitude of the magnetic ordering energy, we show that the disruption of magnetic order as a function of pressure or doping can be qualitatively reproduced but that in calculation, it is achieved through diminishment of $|\mu|$, and therefore likely does not reflect the same physics as detected in experiment. We also find that the strength of the stripe order as a function of doping is strongly site dependent: magnetism decreases monotonically with the number of electrons doped at the Fe site but increases monotonically with the number of electrons doped at the Ba site. Intraplanar magnetic ordering energy (the difference between checkerboard and stripe orderings) and interplanar coupling both follow a similar trend.

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I. INTRODUCTION

The magnetic properties of the Fe-based superconductors are believed to be the key to understanding their normal and superconducting properties. Yet a consensus about the microscopic physics of magnetism in these materials is still lacking. There are several widely held ideas that are arguably supported by most researchers in the field. First, the magnetism is intimately related to the crystal structure, both in terms of the Fe-As bond length, which is reduced when the local magnetic moment on Fe disappears (a simple reflection of the magnetostriuctive nature of Fe) and in terms of an orthorhombic distortion in the magnetically ordered state (though in the 1111 systems, the distortion precedes magnetic ordering in temperature, it is nearly universally believed that the distortion is driven by magnetism and not the other way around). The orthorhombicity of up to 1% is comparable with, say, the rhombohedral distortion of 1.2% in FeO upon the antiferromagnetic (AFM) ordering.

Second, although initial opinions about the origin of the magnetic ordering in Fe pnictides stretched from a spin-Peierls philosophy to Mott physics, it has now been recognized that while the local magnetic moments on Fe are formed independently of the fermiology, their mutual interaction is largely controlled by the itinerant electrons’ response and by the Fermi-surface geometry. A corollary of this fact is that when the long-range order is destroyed (whereupon superconductivity usually emerges), the system should be described as paramagnetic, a collection of disordered magnetic moments, rather than nonmagnetic, with the magnetic moment uniformly suppressed, as in nonpolarized density-functional calculations. Particularly questionable are attempts to describe the evolution of magnetic (and therefore crystallographic) properties when magnetism is suppressed (for instance, by pressure). It has been established that density-functional theory within the generalized gradient approximation (DFT-GGA) describes the crystal structure (as well as the phonon spectra) of the parent compounds very accurately at ambient pressure, as long as full magnetization is allowed. It is not clear, however, whether DFT-GGA will work as well under pressure (the argument above suggests it may not) One purpose of this paper is to address this question.

Another unresolved and important question is the underlying mechanism by which the AFM order is destroyed by external means. Experimentally, one can proceed in three different ways. Chronologically the first method used was formally similar to that used in superconducting cuprates: charge doping. Naturally, it was implicitly assumed that, as in cuprates, charge doping increases the number of carriers, improves the metallic screening and renders the system less strongly interacting, and thus, less magnetic. In accordance with this concept, it was discovered that Ni (which donates two electrons) is about twice more efficient in destroying the long-range magnetism as Co (which donates only one) and that electron doping (substituting O by F, or Fe by Co and Ni) has qualitatively the same effect as hole doping (substituting Ba by K). However, later it was found that pressure and/or strain can lead to essentially the same effect, suggesting that the carrier concentration is not the only, and maybe not even the most important change brought about by the chemical doping. This view was further reinforced by the fact that partial substitution of As by P (which exerts chemical pressure on Fe) has again the same effect. Finally, it was also shown that diluting the Fe plane by nonmagnetic atoms, such as Ru, again destroys the magnetic order and triggers superconductivity.

DFT calculations can account for the last two effects, at least on the qualitative level: physical (volume reduction) or chemical (reducing the iron-pnictogen height) pressure in calculations reduces the tendency to magnetism. However, it is not immediately clear what effect charge doping should
have on magnetism inside DFT. In particular, if the mechanism of suppression is not the same as in cuprates, why would both hole and electron doping have the same, negative effect on magnetism? Answering this question is the second goal of this paper. We find that DFT does show the same qualitative behavior, doping electrons at the Fe site in BaFe$_2$As$_2$ depresses the magnetism, as does doping holes at the Ba site while, intriguingly, doping holes on the Fe site and electrons on the Ba site enhances it.

II. METHODS

All calculations as a function of pressure were carried out using the Vienna \textit{ab initio} simulation package (VASP),\cite{Kresse1996, Kresse1993} a projector augmented wave based pseudopotential formalism. We employed the GGA of Perdew-Burke-Ernzerhof (PBE).\cite{Perdew1996} We fully relaxed a series of structures (both lattice and internal coordinates) at a variety of volumes and extracted the pressure by fitting to an equation of state. All calculations as a function of doping were carried out using WIEN2K,\cite{Blaha2001} which employs an augmented plane wave plus local orbitals basis set, again using PBE-GGA. The lattice coordinates used were $a=5.576$, $b=5.616$, $c=12.950$ Å, and $z_{Fe}=0.8972$ (as a fraction of $c$), corresponding to the fully relaxed structure at zero pressure described previously. These values are in very good agreement with experimental ones\cite{Kimber2010, Joaillier2011, Kang2010, Kim2010, Schneidemann2010} ($a=5.571$, $b=5.615$, and $c=12.970$), in agreement with previous DFT studies done under the same conditions.\cite{Kimber2010, Joaillier2011} To simulate charge doping without using a supercell, we employed the virtual-crystal approximation (VCA). This technique involves replacing each atom of a certain type in the unit cell with a fictitious element with a noninteger atomic number.

For electron doping at the Fe site, we replace $Z=26$ with $Z=26+x$ (toward Co) and for hole doping at the Ba site we use an element with $Z=56-x$ (toward Cs), using the same crystal structure. The number of electrons in the systems is increased commensurately so that overall charge balance is maintained (alternate hole/electron doping at the Co/Ba site is achieved by simply subtracting/adding to $Z$). For calculations of intraplanar and interplanar couplings, we used two separate symmetries, 

\begin{itemize}
  \item Cccm (space group 66) for the observed antiferromagnetically stacked stripe ordering,
  \item Cmma (space group 67) for the ferromagnetically stacked stripe order, and
  \item I4m2 (space group 119) for checkerboard ordering.
\end{itemize}

III. STRUCTURE AS A FUNCTION OF PRESSURE

In Ref. 13, Kimber \textit{et al.} found that both doping and pressure cause the lattice parameters to decrease linearly. The Fe-As bond is found to be extremely rigid, in good agreement with previous experimental study\cite{Joaillier2011} while the As-Fe-As angle changes, whereas in DFT calculations, the Fe-As bond length and As-Fe-As angle (Fig. 2). The former shrinks linearly with pressure in DFT calculations, instead of maintaining the observed constant value. The As-Fe-As angle, on the other hand, is rather constant over the pressure range, whereas in experiment it decreases. Both discrepancies are due to a single factor: the perpendicular height of the As atom above
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the Fe plane (the in-plane component of the Fe-As bond length is determined by \( a \) and \( b \) which both match well with experiment). This height scales linearly with the magnetic moment of the Fe atom, \( \mu \). The physical meaning of this is clear: as discussed, the Fe ion is characterized by a large magnetostrictive effect; compressing the ion results in a loss of the local magnetic moment. The Fe-As bond length controls the chemical pressure on Fe and thus is strongly correlated with the moment. The constant bond length in experiment reveals that the magnitude of the magnetic moment does not change under pressure, indicating that the suppression of magnetic ordering occurs through increased spin fluctuations and orientational disorder rather than through an actual decrease in the absolute magnitude of the moment. DFT does not capture this effect, compensating instead by decreasing the overall moment. The calculated As-Fe-As angle suffers similarly from a decrease in As height that offsets the decrease in \( a \) and \( b \), leaving a relatively constant value.

In view of the fact that the calculated equilibrium moment is larger than the experimentally measured one, one might assume that it would be more rigid than in experiment. The fact that the opposite relationship takes place tells us that while DFT overestimates the ordered moment, it underestimates the local moment. In retrospect, this is not that surprising because there exist residual Coulomb correlations in the system [dynamical mean field theory (DMFT) calculations in the 1111 systems\(^{28} \) indicate about 70\% mass renormalization due to local Coulomb correlation, a small but not negligible number, which enhances the tendency toward local magnetism]\(^{29} \).

IV. MAGNETISM AS A FUNCTION OF PRESSURE

We investigated the interplanar coupling (the total-energy difference between stripe layers stacked antiferromagnetically and ferromagnetically) as a function of pressure. If the coupling were a result of superexchange between Fe layers (whether directly through As-As hopping or through Ba atoms), one would expect it to increase as the layers are pushed closer together. As can be seen in Fig. 3, there is a very slight increase in \( J_\perp \), defined as \( \Delta E = J_\perp \mu^2 \), as the pressure increases, but it is offset by a decrease in the magnetic moment, leaving the net coupling parameter essentially constant (even decreasing very slightly) across the pressure range of 0–6 GPa. In conjunction with the fact that the energy difference between the checkerboard and stripe in-plane magnetic configurations decreases with pressure, these results are again consistent with a picture in which increased spin fluctuations destroy the long-range order. However, as pointed out earlier, the decrease in the magnitude of \( \mu \) as calculated by DFT may not accurately represent reality. It seems more likely that \( |\mu| \) is constant but increasingly fluctuates with pressure. In this case, the interplanar coupling would indeed increase with pressure and the observed suppression of magnetic long-range order must have a different origin, perhaps stemming from in-plane fluctuations.

V. MAGNETISM AS A FUNCTION OF DOPING

One way to gauge the strength of the tendency toward magnetism is to evaluate the energy difference between a magnetic and a nonmagnetic (no local moments) solution. We have calculated this energy difference (Fig. 4) by using the virtual-crystal approximation imitating the Co doping on the Fe site and the K doping on the Ba site (see Sec. II for details). We have further verified (Fig. 4) that supercell calculations for Ba\(_2\)Fe\(_2\)CoAs\(_2\) are quantitatively consistent with the VCA and for BaKFe\(_4\)As\(_4\) semiquantitatively consistent.

Our results show that, in agreement with the experiment, both types of doping weaken the magnetism (reduce the magnetization energy). But, we also found that extending our VCA calculations onto the opposite sides of the phase diagrams, that is, introducing holes on Fe sites or electrons on Ba sites, the trend simply continues so that in these two case the magnetism is enhanced. This same trend was found for a DFT study of the Sr-based 122 compound\(^{26,30} \). Neither of the regimes precisely corresponding to our calculations has been

FIG. 3. (Color online) Interplanar coupling, calculated as the difference in total energies between a system in which the in-plane stripe order is antialigned in successive planes and a system in which in-plane stripe orders are aligned in successive planes.

FIG. 4. (Color online) The magnetic energy, defined as the total energy of the system in the magnetic stripe phase minus the total energy of the nonmagnetic system, as a function of hole and electron doping. Doping on the Fe site with Mns/Co and on the Ba site for K/La are shown for the virtual-crystal approximation. Filled symbols show supercell calculations.

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Fig. 5 (Color online) Interplanar coupling, calculated as in Fig. 3, but as a function of doping rather than pressure. The virtual-crystal approximation is used, with hole doping taking place at the Ba site and electron doping taking place at the Fe site.

Access of the virtual-crystal approximation is used, with hole doping taking place at the Ba site and electron doping taking place at the Fe site.

Fig. 6 (Color online) The magnetic energy difference between in-plane stripe and in-plane checkerboard orderings as a function of doping in the virtual-crystal approximation.

VI. CONCLUSIONS

To summarize, we have extended the familiar DFT-GGA calculations to address several issues not addressed previously. Our findings are as follows:

(1) Although spin-polarized DFT-GGA predicts the equilibrium crystal structure at zero pressure exceedingly well, it becomes increasingly worse with pressure. Specifically, the Fe-As bond is significantly shorter in the calculations than in the experiment. We interpret this as evidence that the local magnetic moment (as opposed to the average ordered moment) is smaller in the calculations, not larger, than in the experiment, and ascribe this to residual local Coulomb correlations.

(2) The antiferromagnetic interlayer coupling is mainly constant as a function of pressure in DFT, which indicates that it is not of pure superexchange origin. We interpret this as an indication of at least two competing interlayer interactions, one antiferromagnetic and one ferromagnetic (double exchange), whose pressure dependencies cancel one another.

(3) The effect of doping strongly depends on the location of the doped charge. Electronic doping in the Fe plane or hole doping in the Ba plane reduces the tendency to form local moments while hole doping in the Fe plane or electron doping in the Ba plane enhances it.

(4) The interplanar coupling is essentially insensitive to doping within conventional doping scheme (holes on the Ba site or electrons on the Fe site).

(5) Intraplanar coupling again shows strong site dependence, but decreases very strongly as a function of doping in either direction within the conventional doping scheme, further supporting the idea that the role of dopants in suppressing magnetism is to increase spin fluctuations.

Note added in proof. Recently, a successful electron doping on the alkaline earth site in a 222 material (La in
SrFe$_2$As$_2$ was reported.\textsuperscript{35} Contrary to the LDA prediction, a suppression of magnetism was observed. However, this first report leaves many questions open; for instance regarding the sample quality.

1. I. I. Mazin and J. Schmalian, in the special issue of Physica C \textbf{469}, 614 (2009).
2. I. I. Mazin, D. J. Singh, M. D. Johannes, and M. H. Du, Phys. Rev. Lett. \textbf{101}, 057003 (2008).
3. V. Chubukov, D. V. Efremov, and I. Eremin, Phys. Rev. B \textbf{78}, 134512 (2008).
4. Q. Si and E. Abrahams, Phys. Rev. Lett. \textbf{101}, 076401 (2008).
5. M. D. Johannes and I. I. Mazin, Phys. Rev. B \textbf{79}, 220510(R) (2009).
6. S. Moon, J. Shin, D. Parker, W. Choi, I. Mazin, Y. Lee, J. Kim, N. Sung, B. Cho, S. Khim, J. Kim, K. Kim, and T. Noh, Phys. Rev. B \textbf{81}, 205114 (2010).
7. T. Yildirim, Phys. Rev. Lett. \textbf{101}, 057010 (2008).
8. Z. P. Yin, S. Lebègue, M. J. Han, B. P. Neal, S. Y. Savrasov, and W. E. Pickett, Phys. Rev. Lett. \textbf{101}, 047001 (2008); K. D. Belashchenko and V. P. Antropov, Phys. Rev. B \textbf{78}, 212505 (2008).
9. I. I. Mazin, M. D. Johannes, L. Boeri, K. Koepernik, and D. J. Singh, Phys. Rev. B \textbf{78}, 085104 (2008).
10. A. Jesche, N. Caroca-Canales, H. Rosner, H. Bormann, A. Ormeci, D. Kasinathan, H. H. Klaus, H. Luetkens, R. Khasanov, A. Amato, A. Hoser, K. Kaneko, C. Krellner, and C. Geibel, Phys. Rev. B \textbf{78}, 180504(R) (2008).
11. P. C. Canfield, S. L. Bud’ko, J. Q. Ni Ni, Yan, and A. Kracher, Phys. Rev. B \textbf{80}, 060501 (2009).
12. P. L. Alireza, Y. T. C. Ko, J. Gillet, C. M. Petrone, S. E. Sebastian, and G. G. Lonzarich, J. Phys.: Condens. Matter \textbf{21}, 012208 (2009).
13. S. A. J. Kimber, A. Kreysig, Y.-Z. Zhang, H. O. Jeschke, R. Valenti, F. Yokaichiya, E. Colombier, J. Yan, T. C. Hansen, T. Chatterji, R. J. McQueeney, P. C. Canfield, A. I. Goldman, and D. N. Argyriou, Nature Mater. \textbf{8}, 471 (2009).
14. Z. Ren, Q. Tao, S. Jiang, C. Feng, C. Wang, J. Dai, G. Cao, and Z. Xu, Phys. Rev. Lett. \textbf{102}, 137002 (2009).
15. S. Sharma, A. Bharathi, S. Chandra, R. Reddy, S. Paulraj, A. Satya, V. Shastry, A. Gupta, and C. Sundar, Phys. Rev. B \textbf{81}, 174512 (2010).
16. M. A. McGuire, D. J. Singh, A. S. Sefat, B. C. Sales, and D. Mandrus, J. Solid State Chem. \textbf{182}, 2326 (2009).
17. G. Kresse and J. Furthmuller, Phys. Rev. B \textbf{54}, 11169 (1996).
18. P. E. Blöchl, Phys. Rev. B \textbf{50}, 17953 (1994).
19. J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. \textbf{77}, 3865 (1996).
20. P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnicka, and J. Luitz, WIEN2K, An augmented Plane Wave plus Local Orbitals Program for Calculating Crystal Properties (Technische Universität Wien, Austria, 2002).
21. M. Kofu, Y. Qiu, W. Bao, S.-H. Lee, S. Chang, T. Wu, G. Wu, and X. H. Chen, New J. Phys. \textbf{11}, 055001 (2009).
22. M. Zbiri, H. Schober, M. R. Johnson, S. Rols, R. Mittal, Y. Su, M. Rotter, and D. Johrendt, Phys. Rev. B \textbf{79}, 064511 (2009).
23. Q. Huang, Y. Qiu, W. Bao, M. A. Green, J. W. Lynn, Y. C. Gasparovic, T. Wu, G. Wu, and X. H. Chen, Phys. Rev. Lett. \textbf{101}, 257003 (2008).
24. M. Rotter, M. Tegel, D. Johrendt, I. Schellenberg, W. Hermes, and R. Potting, Phys. Rev. B \textbf{78}, 020503(R) (2008).
25. Y.-Z. Zhang, H. C. Kandpal, I. Opahle, H. O. Jeschke, and R. Valenti, Phys. Rev. B \textbf{80}, 094530 (2009).
26. D. Kasinathan, A. Ormeci, K. Koch, U. Burkhardt, W. Schnelle, A. Leithe-Jasper, and H. Rosner, New J. Phys. \textbf{11}, 025023 (2009).
27. M. Rotter, M. Pangerl, M. Tegel, and D. Johrendt, Angew. Chem. \textbf{47}, 7949 (2008).
28. M. Aichhorn, L. Pourovskii, V. Vildosola, M. Ferrero, O. Parcollet, T. Miyake, A. Georges, and S. Biermann, Phys. Rev. B \textbf{80}, 085101 (2009).
29. A. G. Petukhov, I. I. Mazin, L. Chioncel, and A. I. Lichtenstein, Phys. Rev. B \textbf{67}, 153106 (2003).
30. Y. Kim, S. Khim, H. Kim, M. Eom, J. Law, R. Kremer, J. Shim, and K. Kim, Phys. Rev. B \textbf{82}, 024510 (2010).
31. N. Pérez, P. Guardia, A. Roca, M. P. Morales, C. Serna, O. Iglesias, F. Bartolome, L. Garcia, X. Batlle, and A. Labarta, Nanotechnology \textbf{19}, 475704 (2008).
32. T. Yildirim, Physica C \textbf{469}, 425 (2009).
33. A. Yaresko, G. Liu, V. Antonov, and O. Andersen, Phys. Rev. B \textbf{79}, 144421 (2009).
34. L. W. Harriger, A. Schneidewind, S. Li, J. Zhao, Z. Li, W. Lu, X. Dong, F. Zhou, Z. Zhao, J. Hu, and P. Dai, Phys. Rev. Lett. \textbf{103}, 087005 (2009).
35. Y. Muraba, S. Matsuishi, S.-W. Kim, T. Atou, O. Fukunaga, and H. Hosono, arXiv:1005.0528 (unpublished).

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