Doping-Induced Dielectric Catastrophe Prompts Free-Carrier Release in Organic Semiconductors
Massimiliano Comin, Simone Fratini, Xavier Blase, Gabriele d’Avino

To cite this version:
Massimiliano Comin, Simone Fratini, Xavier Blase, Gabriele d’Avino. Doping-Induced Dielectric Catastrophe Prompts Free-Carrier Release in Organic Semiconductors. Advanced Materials, 2022, 34 (2), pp.2105376. 10.1002/adma.202105376. hal-03411943

HAL Id: hal-03411943
https://hal.science/hal-03411943
Submitted on 2 Nov 2021

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L’archive ouverte pluridisciplinaire HAL, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d’enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.
Doping-induced dielectric catastrophe prompts free-carrier release in organic semiconductors

Massimiliano Comin Simone Fratini Xavier Blase Gabriele D’Avino*

Grenoble Alpes University, CNRS, Grenoble INP, Institut Néel, 25 rue des Martyrs, 38042 Grenoble, France
*gabriele.davino@neel.cnrs.fr

Keywords: doping, organic semiconductors, dielectric screening, charge transfer, polarizability, QM/MM, dielectric catastrophe, insulator-conductor transition

The control over material properties attainable through molecular doping is essential to many technological applications of organic semiconductors, such as OLED or thermoelectrics. These excitonic semiconductors typically reach the degenerate limit only at impurity concentrations of 5-10%, a phenomenon that has been put in relation with the strong Coulomb binding between charge carriers and ionized dopants, and whose comprehension remained elusive so far. This study proposes a general mechanism for the release of carriers at finite doping in terms of collective screening phenomena. A multiscale model for the dielectric properties of doped organic semiconductor is set up by combining first principles and microelectrostatic calculations. Our results predict a large nonlinear enhancement of the dielectric constant (ten-fold at 8% load) as the system approaches a dielectric instability (catastrophe) upon increasing doping. This can be attributed to the presence of highly polarizable host-dopant complexes, plus a nontrivial leading contribution from dipolar interactions in the disordered and heterogeneous system. The enhanced screening in the material drastically reduces the (free) energy barriers for electron-hole separation, rationalizing the possibility for thermal charge release. The proposed mechanism is consistent with conductivity data and sets the basis for achieving higher conductivities at lower doping loads.

1 Introduction

The ability to control the charge carrier density and transport levels is key to the success of semiconducting technologies and organic semiconductors (OSC)s made of π-conjugated molecules or polymers make no exception. Chemical doping by means of electron attracting or withdrawing impurity molecules emerged as a successful approach that lies at the foundation of stable and efficient organic optoelectronic and thermoelectric devices.[1, 2, 3, 4] While recent years have witnessed remarkable advances concerning the chemistry[5, 6, 7] and the structural control[8, 9, 10, 11] of doped OSCs, the microscopic phenomena that govern molecular doping remain to date unclear.

A central open question concerns the release of free doping-induced charge carriers in low-dielectric constant organic materials. Experimental data for small-molecules and polymers show that a boost of the electrical conductivity by orders of magnitude is typically achieved at doping loads of about 5-10%, i.e. at concentrations that are orders of magnitude higher than in inorganic semiconductors.[12, 13, 14] Several studies have shown that the conductivity follows a thermally activated behavior,[15, 16, 17, 18] and the activation energy has been related to the Coulomb interaction between an ionized dopant and the charge injected in the semiconductor.[18] Photoemission measurements[19] and theoretical calculations[20, 21, 22] set this binding-energy in the 400-700 meV range in the low-doping regime, a value that is too large to permit a significant release of free carriers at room temperature.

Experimental data for p and n-doped OSCs have shown a universal tendency for a conspicuous reduction of the activation energy with dopant concentration.[18, 16, 17, 23] Besides, the conductivities of a set of polymers heavily-doped by ion-exchange have been shown to be independent on the ion size,[24] which is at odds with the interpretation of transport limited by Coulomb interactions. This set of evidences, together with the high loads needed to boost conductivity, points to a collective depinning mechanism for charge carriers taking place upon increasing doping concentration, beyond trap filling effects at ultralow doping.[17, 25]
Theoretical studies mostly focused on the infinite dilution limit, discussing single host-dopant complexes with quantum or hybrid quantum/classical (QM/MM) methods.[26, 13, 19, 27, 22] Some of us applied embedded many-body perturbation techniques and an essential-state model for charge transfer (CT) degrees of freedom to the paradigmatic case of F4TCNQ-doped pentacene,[28, 12, 26] showing that the full dopant ionization takes place at room-temperature thanks to the excitonic e-h interaction and structural relaxation. That study predicted also the occurrence of very low-energy (down to 34 meV) CT excitations in spite of acceptor levels being very deep in the host gap. Coming to transport, Kinetic Monte Carlo simulations proved able to reproduce the doping-induced conductivity enhancements observed in experiments.[29, 30, 31, 32] Those results have been interpreted in terms of the favorable effect of the energetic disorder sourced from the Coulomb field of doping-induced charges, assisting charge carriers’ release through the formation of energetically favorable pathways.[33]

In this article, we investigate the effect of collective screening phenomena on the release of doping-induced charges. Following a multiscale approach, we first evaluate \textit{ab initio} the polarizability associated with low-energy CT degrees of freedom in host-dopant complexes, showing that this contribution is as large as ten times that of a single molecule, regardless of the fractional or integer CT nature of the complex ground state. This information is then used to build a microscopic model for doped OSCs that allows us to obtain the dielectric properties of doped OSCs as a function of the impurity concentration. Our calculations reveal a large enhancement of the relative permittivity $\varepsilon_r = \varepsilon/\varepsilon_0$ upon doping, with an order of magnitude increase at 8% concentration, which implies a drastic suppression of Coulomb energy barriers for the release of free charge carriers. The origin of this enhancement is rooted in the collective response of highly-polarizable host-dopant complexes dopants in a disordered system, whose susceptibility increases upon doping and diverges at the approach of the dielectric catastrophe.

2 Results

Our analysis starts from first-principles calculations of the polarizability of host-dopant complexes and considers the paradigmatic case of a F4TCNQ impurity in the pentacene crystal, see Fig. 1a. This system is described with hybrid quantum/classical (QM/MM) calculations, where we treat the complex composed by the dopant and its first shell of six pentacene neighbors at the density functional theory (DFT) level. Considering the primary importance of a correct description of excitation energies for polarizabilities, and the strong variability of energy levels with DFT functionals, our calculations employed a hybrid functional tuned to reproduce the gap obtained from accurate embedded $GW$ calculations (see Methods and Supporting Information).[20] The surrounding crystalline environment is described by an atomistic polarizable model.[34, 35]

Our results in Table 1 reveal that the polarizability of a host-dopant complex in the gas-phase is significantly larger than the sum of the molecular polarizabilities, testifying a contribution from intermolecular CT interactions. Moreover, the polarizability of the complex roughly doubles when accounting for the MM environment, an observation that can be equivalently explained in terms of the screening of the dipolar field generated by the induced charge density ($\Delta n$, shown in Figure 1a), or as a result of the stabilization of CT excitations by the polarizable environment. The full QM/MM polarizability of the host-dopant complex, $\alpha_{cpx} = 804 \, \text{Å}^3$, accounts for both an intramolecular an intermolecular CT contribution. Upon
**Table 1:** Polarizability of pentacene and F4TCNQ molecules and their complex shown in Figure 1a, calculated at DFT level in the gas and in the solid state (QM/MM embedding). The last two lines report the polarizability associated to intermolecular CT degrees of freedom, calculated at QM/MM level and with a generalized Mulliken model for intermolecular CT. Reported polarizabilities correspond to the average of the tensor components in the pentacene crystal’s $ab$ plane, see Figure 1a. Note that the CT polarizability of the pentacene-F4TCNQ is one order of magnitude larger than that of a single molecule.

| System                     | Method      | $\alpha$ (Å$^3$) |
|----------------------------|-------------|------------------|
| Pentacene                  | DFT gas     | 47               |
| F4TCNQ                     | DFT gas     | 33               |
| F4TCNQ-pentacene           | DFT gas     | 418              |
| F4TCNQ-pentacene           | DFT/MM      | 804              |
| F4TCNQ-pentacene, CT       | DFT/MM      | 534              |
| F4TCNQ-pentacene, CT       | CT model    | 357              |

Dissecting the two terms by means of an approximate subtractive scheme, we have been able to quantify a leading CT contribution $\alpha_{CT} = 534$ Å$^3$, which is above ten times that of a single pentacene molecule. Full detail on polarizability calculations are provided as SI.

Such a remarkably large value compares well with the estimate from a simpler Mulliken model for intermolecular CT (see Table 1), which has been carefully parameterized for F4TCNQ-doped pentacene and validated against embedded Bethe-Salpeter Equation calculations.[20] As shown in detail in the SI, model calculations allowed us to extend and generalize our first-principles findings to other systems, including a realistic amorphous sample of F6TCNNQ-doped NPB.[21] The CT polarizabilities of complexes extracted from this amorphous system present an order of magnitude variability, with average of 577 Å$^3$ and values as large as 1000 Å$^3$. Finally, in order to establish our findings on very general grounds, we have performed Mulliken model calculations spanning the set of microscopic parameters (i.e. gap, CT integrals and energetic disorder) relevant to doped OSCs. This analysis allowed us to conclude that host-dopant complexes featuring both a fractional and an integer CT in the ground state are intrinsically highly polarizable (see SI Figures S3 and S4). Indeed, in these two scenarios which have been documented in experiments,[13] the excitation spectrum of host-dopant complexes is characterized by the presence of dipole-allowed CT
transitions at very low energy that largely contribute to the polarizability.

Having characterized the polarizabilities associated with host-dopant complexes, we are in a position to set up a model for the dielectric properties of OSCs at finite doping load. As a first step towards a more realistic description, we describe this heterogeneous dielectric medium with an induced-dipole scheme on a face-centered cubic (FCC) lattice, whose sites represent either a host molecule or a host-dopant complex, see Fig. 1b,c. Dopants are randomly distributed over the lattice and a polarizability $\alpha_{CT} = 500 \text{ Å}^3$ is assigned to these sites. The polarizability of host sites is set to $\alpha_{host} = 50 \text{ Å}^3$, which corresponds to that of a pentacene molecule. This, for the lattice constant of 12.26 Å used throughout this work, corresponds to a dielectric constant of $\varepsilon_r = 3.5$ for the pristine host, according to the standard Clausius-Mossotti relation. We note that while the dielectric properties and the related dipole-field sums have been extensively investigated in high-symmetry lattices of equivalent sites,[36, 37, 38, 39] to the best of our knowledge, the effect of disorder and inhomogeneity remains completely unexplored to date.

Following the common treatments of dipolar linear response, the dipoles induced in the system by the electric field of permanent sources $F_0$ (e.g. an external field or a point charge inside the material) can be obtained by solving the linear system

$$H\mu = F_0$$

where $\mu$ and $F_0$ are vectors of the Cartesian components of induced dipoles and permanent fields at the lattice sites. The Hessian matrix $H$ consists of the second derivatives of the system’s energy with respect to the interacting dipoles, accounting for site polarizability and dipole-dipole interactions. The advantage of this approach consists in the possibility to describe with numerically exact calculations inhomogeneous systems of thousands of molecular sites, possibly accounting for periodic boundary conditions: as a result, the present method allows to span the interval of doping loads of relevance for experiments.

A first insight on the effect of the increasing dopant density, $\rho$, can be obtained from the spectrum of the Hessian matrix of an extended system. The Hessian eigenvalues $\lambda$ quantify the stiffness of the collective polarization modes of the system of interacting polarizabilities, i.e. their inverse, $\lambda^{-1}$, is proportional to the polarizability associated to each normal mode. In particular, vanishing positive Hessian eigenvalues determine a finite polarization even in the presence of a tiny perturbation, while modes characterized by $\lambda < 0$ are unstable, i.e. develop a spontaneous polarization even in the absence of an electric field.

The distribution of $\lambda$ values at different doping loads is shown in Figure 2, where the vertical lines mark $\alpha_{host}^{-1}$ and $\alpha_{CT}^{-1}$, i.e. the stiffness of non-interacting host and dopant sites, respectively. A sizable eigenvalue dispersion due to inter-site interactions is obtained already in the pristine system, $\rho=0\%$. This distribution is bimodal, with a peak at $13 \cdot 10^{-3} \text{ Å}^{-3}$ (transverse polarization modes branch), and a much broader feature centered around $30 \cdot 10^{-3} \text{ Å}^{-3}$ (longitudinal modes). Upon increasing doping, the shape of the distribution barely changes in the region corresponding to the host sites ($\lambda > 10 \cdot 10^{-3} \text{ Å}^{-3}$). On the other hand, new peaks resulting from highly polarizable host-dopant complexes appear around $\alpha_{CT}^{-1}$ (dashed vertical line). These peaks grow in intensity and significantly broaden with increasing concentration as a result of the interaction between complexes. Most interestingly, the low-$\lambda$ tail of the distribution approaches zero upon increasing $\rho$, with 1.6% of the modes associated to host-dopant complexes becoming unstable ($\lambda < 0$) at 8% doping. This softening of the collective polarization modes with doping signals an enhancement of the susceptibility as the system moves toward a dielectric catastrophe.[40]

We next turn our attention to the bulk dielectric constant, that we compute by generalizing the approach
Figure 2: Distributions of the Hessian matrix eigenvalues, measuring the stiffness of the collective polarization modes in a doped OSC, for increasing impurity concentration $\rho$. Results obtained with an induced-dipole model on a lattice, with site polarizabilities corresponding to the typical values for doped OSCs, e.g. pentacene-F4TCNQ (see Table 1). The vertical lines mark the inverse polarizabilities of non-interacting host molecules and host-dopant complexes. The distributions are characterized by two main features associated to the modes of host sites ($\lambda > 10 \cdot 10^{-3}$ Å$^{-3}$) and to complexes ($\lambda < 4 \cdot 10^{-3}$ Å$^{-3}$). The latter features an increasing dispersion with $\rho$, with the low-$\lambda$ tail of the distribution approaching, and eventually surpassing, zero. This interaction-driven softening results in an enhanced susceptibility to perturbations. The y-scale axis in the left panel has been magnified for better visualization.

Our calculations predict a divergent behavior of the dielectric constant, leading to a large non-linear enhancement as function of the doping load, shown by the black dots in Figure 3. This amounts to a four (ten) fold increase of $\varepsilon_r$ at 6% (8%) doping with respect to the pristine OSC, which can have a dramatic effect on the release of Coulombically bound charges, as discussed later. We emphasize that such a striking increase of $\varepsilon_r$ occurs at the typical concentrations leading to a boost in the measured electrical conductivity.[1, 3, 2]

The enhancement in the dielectric constant can be rationalized by looking at the polarization response to the external field, $\zeta$. The distributions of $\zeta$ values obtained upon sampling the dopants’ positional disorder are shown as an inset in Fig. 3. These distributions are symmetric, nicely approximated by Gaussians (black dotted lines) up to $\rho = 8\%$, and develop some skewness at larger doping, i.e. when the
dielectric becomes more and more unstable. According to Eq. 2, the dielectric constant diverges (dielectric catastrophe) at $\zeta = 3$ (horizontal dashed line). This determines a remarkable, non-linear amplification of $\varepsilon_r$ upon approaching the singularity from $\zeta < 3$, and negative values afterwards, the latter marking the instability. This explains the histogram distributions of $\varepsilon_r$, shown in the main panel of Figure 3, which shift to higher values, broaden and become more and more skewed upon approaching the dielectric catastrophe by increasing $\rho$. This behaviour is captured to a good approximation by evaluating analytically the distribution of $\varepsilon_r(\zeta)$ (dashed lines, main panel) via Eq. 2, using the Gaussian fits of the $\zeta$ distributions.

Figure 3: Enhancement of the bulk dielectric constant of a doped organic semiconductor with the impurity concentration $\rho$. Results obtained with an induced-dipole model on a lattice, with site polarizabilities corresponding to the typical values for doped OSCs, e.g. pentacene-F4TCNQ (see Table 1). The main panel shows the distribution (histograms) and harmonic mean (dots) of $\varepsilon_r$ obtained upon sampling over dopants positional disorder. The full line corresponds to the dielectric constant obtained for an effective homogeneous medium with mean site polarizability $\alpha_{\text{avg}}(\rho)$, see text. The inset shows the dependence of the external-field susceptibility $\zeta$ on doping, displaying numerical results for the inhomogeneous lattice (distributions and their Gaussian fits as dashed lines) and $\zeta(\rho)$ for an effective homogeneous medium (full line). The Gaussian fits are then analytically transformed into distributions for $\varepsilon_r$, shown as dashed lines in the main panel. The right-hand panel illustrates the system behavior at the dielectric catastrophe, note the different scale on the ordinate.

The picture that emerges from our results is that two phenomena cooperate in determining the boost of the dielectric constant with $\rho$. The first one is the straightforward effect of increasing the concentration of highly polarizable host-dopant complexes. This effect can be captured by considering an effective medium with average site polarizability $\alpha_{\text{avg}}(\rho) = (1 - \rho)\alpha_{\text{host}} + \rho\alpha_{\text{CT}}$. This effective medium approximation, shown as a solid line in Figure 3, only accounts for 26% of the doping-induced enhancement of $\varepsilon_r(\rho = 8\%)$ (dots). The remaining, dominant, contribution originates from cooperative dipolar interactions taking place in the disordered lattice, which result in microscopic dipole fields that are strongly polarizing (i.e. enhancing the external field) at highly-polarizable dopant sites (see SI, Figure S3). As a result, the external-field susceptibility $\zeta$ presents a faster increase with doping in the inhomogeneous than in the
effective homogeneous medium (inset of Figure 3, full line), attaining the divergence at a lower doping. This marks a crucial qualitative difference between a homogeneous dielectric, which obeys the Clausius-Mossotti equation, and an intrinsically chemically inhomogeneous disordered medium, such as a doped OSC.

Finally, we discuss the energetics of charge dissociation in the presence of the enhanced dielectric screening sourced from highly-polarizable host-dopant complexes. To such an aim, we calculate the collective response to the field of a point-charge $F_0 = F_q$ (see Figure 1c), which gives us access to charge separation energy profiles as a function of the electron-hole distance $r_{eh}$, shown in panels a-c of Figure 4. In the infinite dilution limit ($\rho = 0$, panel a), except for small oscillations at short distance, the energy profile (dots) has the form of a Coulomb potential, screened by the bulk $\varepsilon_r = 3.5$ (full line), similar to analogous atomistic results.[20] The energy barrier to free charges sitting on nearest-neighbour molecules ($E_b$ in Figure 4a-c) amounts to 0.5 eV, consistent with first-principles calculations [20, 19] and conductivity data at low doping.[18]

The situation radically changes at finite dopants concentration. First, the screening provided by host-dopant complexes determines a sizable suppression of the average charge-separation energy barrier, reducing to 0.29 and 0.15 eV at 4 and 8% doping, respectively. We anticipate that this large screening effect provides a decisive contribution to the charge release mechanism explained below. Second, our calculations provide insights into the effects of the inhomogeneity of the medium. This determines a substantial spread in the energy profile for charge separation (gray shaded areas in Figure 4b,c), with important local deviations from the screened Coulomb potential with bulk $\varepsilon_r$, which is recovered only at large $r_{eh}$. The spatial inhomogeneity of the system can be best appreciated from the fluctuations of the non-local microscopic dielectric constants $\varepsilon_r(r, r')$, shown in the insets. These fluctuations, which would be missed by reasoning only in terms of the bulk $\varepsilon_r$, may determine pathways for charge separation that are energetically more favorable than the average one.

In order to ultimately address the question of charge carriers release at room temperature, we now assess the free energy profile for charge separation, shown in Figure 4d-f. The free energy accounts for the entropy contribution resulting from the radial density of microstates, increasing as $r_{eh}^2$ in three dimensions.[43] The entropy variation always assists charge separation, leading to a further decrease of the free-energy barrier ($F_b$ in Figure 4d-f), which reduces to 0.17 and 0.13 eV at 4 and 8% doping, rationalizing the possibility for the thermal release of free carriers at the impurity concentrations that actually correspond to the conductivity boosts seen in experiments.

3 Discussion and Conclusion

We have proposed a model description of collective screening phenomena in doped organic semiconductors and their implications on the release of Coulombically bound charges introduced upon doping. Our treatment based on local polarizabilities associated to highly-polarizable host-dopant complexes and host molecules, which strictly holds upon approaching conducting states from the insulating side (i.e. for bound charges), predicts a factor ten increase of the bulk dielectric constant at doping loads of 8% and very large fluctuations at the microscopic scale. This enhanced screening, together with entropic effects, drastically reduces the Coulomb barrier for charge separation, providing a mechanistic explanation for the impurity
Figure 4: Energy and free-energy profiles for charge separation, showing the large reduction of the barrier for free carrier release ($E_b$ and $F_b$) upon increasing the doping load, as a result of dielectric screening and entropy. (a-c) Energy profiles, mean value (dots) and standard deviation (shaded areas) for the different coordination shells sampled over dopants disorder. The full line corresponds to a screened Coulomb potential calculated with bulk $\varepsilon_r$. The insets show the non-local dielectric constant $\varepsilon_r(\mathbf{r}, \mathbf{r}')$, computed for a source charge at $\mathbf{r}'$ (central site) for a given realization of dopants position. Circles and triangles represent host and dopant sites, respectively. (d-e) Room-temperature free energy profiles, partitioned into energetic and entropic contributions (semi-logarithmic scale). Dots correspond to numerical results, lines are analytic expressions for the continuum.
concentration dependence of the activation energies extracted from conductivity data.[18]

Divergent dielectric responses are *de facto* expected in conducting states and arguments similar to ours, i.e. linking the polarizability of hydrogen-like impurities to $\varepsilon_r$, have been proposed for doped silicon.[44, 45, 46] We remark that the microscopic treatment proposed herein goes far beyond early works based on local-field corrections,[44, 45] as it is able to capture the major role played by inhomogeneity and disorder at the molecular scale.

The present analysis builds a fresh picture of the doping-induced insulator-to-conductor transition in terms of an incipient dielectric catastrophe, that is broadly consistent with available experimental data for organic semiconductors, including the 5-10% loads needed to boost conductivity.[1, 3, 2] Our findings also agree with the conductivity enhancements upon doping observed in earlier studies based on kinetic Monte Carlo simulations [29, 30, 32], yet providing additional microscopic insights on the charge release mechanism. In fact, our analysis indicate that the fundamental cause of the phenomenon lies in many-body screening phenomena, rather that in the direct effect of the energetic disorder sourced from the dipolar fields of host-dopant complexes, as it was previously assumed. Interestingly, the screening effects revealed here are in principle also captured in kinetic Monte Carlo simulations, where they could be specifically accessed by evaluating the space and time correlations of particles trajectories and of the electric polarization, which have not been reported to date. In the prevailing view, charge release is considered the result of the formation of energetically favorable pathways in the rough electrostatic field of many ionized host-dopant complexes.[33, 29, 18] In a strongly interacting environment such as the doped systems studied here, the formation of such energetically favorable pathways can only occur if the Coulomb binding of a probe electron-hole pair is counteracted by the electrical potentials of the other moving charges. The charge rearrangements that make this possible are, in essence, the collective screening processes studied here.

We are confident this work will stimulate experimental studies targeting the observation of low-energy electronic excitations responsible for the large polarizability of host-dopant complexes, as well as the direct measurement of dielectric constant of doped organic semiconductors. A thorough understanding and management of the dielectric properties of doped organic semiconductors represents a promising gateway to the achievement of higher conductivities at lower doping concentration.

4 Methods

QM/MM calculations were performed at the DFT PBEh($a_{HFX}$=0.4)/cc-pVTZ level with the atomistic polarizable embedding scheme described in Ref. [34]. The DFT/MM polarizability is calculated as numerical derivatives of the QM-region dipole induced by an electric field, accounting for the effect of the polarizable MM environment. QM/MM calculations have been performed with the ORCA v4.2 and MESCal[35] codes.

The induced-dipole model for doped OSCs considered a FCC lattice, with dopants randomly distributed over sites with the constraint of not having first and second nearest neighbor impurities. Results have been obtained upon sampling over 1000 random realizations of the distribution of dopants over lattice sites (dopants' positional disorder). The latter actually bear the polarizability of the host-dopant CT complex, hence the interaction between its polarizability and that of the first host neighbors has been neglected. The linear system in Eq. 1 has been solved by full diagonalization or by matrix inversion (Cholesky factorization). Bulk dielectric constant calculations were performed accounting for periodic
boundary conditions in three dimensions. Charge-separation radial energy profiles have been calculated by placing a source charge at $r'$, and computing the energy of a probe charge $E(r) = eW(r, r')$, where $W$ is the electrostatic potential screened by induced dipoles. These calculations have been performed in open-boundary conditions, carefully extrapolating $W$ in the infinite system limit. The nonlocal dielectric constant is obtained as $\varepsilon(r, r') = v(r, r')/W(r, r')$, where $v(r, r') = |r - r'|^{-1}$ is the bare Coulomb potential. Additional computational details are provided as Supporting Information.

Supporting Information
Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements
The authors thank David Beljonne and Henning Sirringhaus for stimulating discussion about doping in organic semiconductors. High performance computing resources from GENCI-TGCC (grant no. 2020-A0090910016) are acknowledged. M.C. acknowledges PhD scholarship funding from Grenoble Quantum Engineering (GreQuE) and Fondation Nanosciences as well as the European Union’s Horizon 2020 research and innovation programme under the Marie Skłodowska-Curie grant agreement No 754303 (GreQue).

References

[1] K. Walzer, B. Maennig, M. Pfeiffer, K. Leo, Chem. Rev. 2007, 107, 4 1233.

[2] I. Jacobs, A. Moulé, Adv. Mater. 2017, 29 1703063.

[3] I. Salzmann, G. Heimel, M. Oehzelt, S. Winkler, N. Koch, Acc. Chem. Res. 2016, 49, 3 370.

[4] W. Zhao, J. Ding, Y. Zou, C.-a. Di, D. Zhu, Chem. Soc. Rev. 2020, 7210–7228.

[5] X. Lin, B. Wegner, K. M. Lee, M. A. Fusella, F. Zhang, K. Moudgil, B. P. Rand, S. Barlow, S. R. Marder, N. Koch, A. Kahn, Nat. Mater. 2017, 16, 12 1209.

[6] Y. Yamashita, J. Tsurumi, M. Ohno, R. Fujimoto, S. Kumagai, T. Kurosawa, T. Okamoto, J. Takeya, S. Watanabe, Nature 2019, 572, 7771 634.

[7] B. Yurash, D. X. Cao, V. V. Brus, D. Leifert, M. Wang, A. Dixon, M. Seifrid, A. E. Mansour, D. Lungwitz, T. Liu, P. J. Santiago, K. R. Graham, N. Koch, G. C. Bazan, T.-Q. Nguyen, Nat. Mater. 2019, 18, 12 1327.

[8] K. Kang, S. Watanabe, K. Broch, A. Sepe, A. Brown, I. Nasrallah, M. Nikolka, Z. Fei, M. Heeney, D. Matsumoto, Nat. mater. 2016, 15, 8 896.

[9] H. Hase, K. O’Neill, J. Frisch, A. Opitz, N. Koch, I. Salzmann, The Journal of Physical Chemistry C 2018, 122, 45 25893.

[10] I. E. Jacobs, C. Cendra, T. F. Harrelson, Z. I. Bedolla Valdez, R. Faller, A. Salleo, A. J. Moulé, Mater. Horiz. 2018, 5 655.

[11] R. Warren, A. Privitera, P. Kaienburg, A. E. Lauritzen, O. Thimm, J. Nelson, M. K. Riede, Nat. Commun. 2019, 10, 1.
[12] H. Kleemann, C. Schuenemann, A. A. Zakhidov, M. Riede, B. Lüssem, K. Leo, *Org. Electron.* **2012**, *13*, 1 58 .

[13] H. Mendez, G. Heimel, S. Winkler, J. Frisch, A. Opitz, K. Sauer, B. Wegner, M. Oehzelt, C. Rothel, S. Duhm, D. Tobbens, N. Koch, I. Salzmann, *Nat. Commun.* **2015**, *6* 8560.

[14] P. Pingel, D. Neher, *Physical Review B* **2013**, *87* 115209.

[15] B. Maennig, M. Pfeiffer, A. Nollau, X. Zhou, K. Leo, P. Simon, *Phys. Rev. B* **2001**, *64* 195208.

[16] Y. Shen, K. Diest, M. H. Wong, B. R. Hsieh, D. H. Dunlap, G. G. Malliaras, *Phys. Rev. B* **2003**, *68* 081204.

[17] S. Olthof, S. Mehraeen, S. K. Mohapatra, S. Barlow, V. Coropceanu, J.-L. Brédas, S. R. Marder, A. Kahn, *Phys. Rev. Lett.* **2012**, *109* 176601.

[18] M. Schwarze, C. Gaul, R. Scholz, F. Bussolotti, A. Hofacker, K. S. Schellhammer, B. Nell, B. D. Naab, Z. Bao, D. Spoltore, K. Vandewal, J. Widmer, S. Kera, N. Ueno, F. Ortmann, K. Leo, *Nat. Mater.* **2019**, *18* 242.

[19] C. Gaul, S. Hutsch, M. Schwarze, K. S. Schellhammer, F. Bussolotti, S. Kera, G. Cuniberti, K. Leo, F. Ortmann, *Nat. Mater.* **2018**, *17* 439.

[20] J. Li, G. D’Avino, A. Pershin, D. Jacquemin, I. Duchemin, D. Beljonne, X. Blase, *Phys. Rev. Mater.* **2017**, *1* 025602.

[21] J. Li, I. Duchemin, O. M. Roscioni, P. Friederich, M. Anderson, E. D. Como, G. Kociok-Köhn, W. Wenzel, C. Zannoni, D. Beljonne, X. Blase, G. D’Avino, *Mater. Horiz.* **2019**, *6* 107.

[22] A. Privitera, G. Lendi, M. Riede, G. D’Avino, D. Beljonne, *Adv. Funct. Mater.* **2020**, *30*, 45 2004600.

[23] F. Zhang, A. Kahn, *Advanced Functional Materials* **2018**, *28*, 1 1703780.

[24] I. E. Jacobs, G. D’Avino, Y. Lin, V. Lemaur, Y. Huang, X. Ren, D. Simatos, W. Wood, C. Chen, T. Harrelson, T. Mustafa, C. A. O’Keefe, L. Spalek, D. Tjhe, M. Statz, L. Lai, P. A. Finn, W. G. Neal, J. Strzalka, C. B. Nielsen, J.-K. Lee, S. Barlow, S. R. Marder, I. McCulloch, S. Fratini, D. Beljonne, H. Sirringhaus, *arXiv* **2021**, 2101.01714.

[25] M. L. Tietze, P. Pahner, K. Schmidt, K. Leo, B. Lüssem, *Adv. Funct. Mater.* **2015**, *25*, 18 2701.

[26] I. Salzmann, G. Heimel, S. Duhm, M. Oehzelt, P. Pingel, B. M. George, A. Schnegg, K. Lips, R. l.-P. Blum, A. Vollmer, N. Koch, *Phys. Rev. Lett.* **2012**, *108* 035502.

[27] A. M. Valencia, C. Cocchi, *J. Phys. Chem. C* **2019**, *123*, 14 9617.

[28] S. D. Ha, A. Kahn, *Phys. Rev. B* **2009**, *80* 195410.

[29] M. L. Tietze, J. Benduhn, P. Pahner, B. Nell, M. Schwarze, H. Kleemann, M. Krammer, K. Zojer, K. Vandewal, K. Leo, *Nat. Commun.* **2018**, *9* 1182.

[30] A. Fediai, F. Symalla, P. Friederich, W. Wenzel, *Nat. Commun.* **2019**, *10* 4547.
[31] A. Fediai, A. Emering, F. Symalla, W. Wenzel, *Phys. Chem. Chem. Phys.* **2020**, *22* 10256.

[32] M. Koopmans, M. A. T. Leiviskä, J. Liu, J. Dong, L. Qiu, J. C. Hummelen, G. Portale, M. C. Heiber, L. J. A. Koster, *ACS Applied Materials & Interfaces* **2020**, *12*, 50 56222.

[33] A. Mityashin, Y. Olivier, T. Van Regemorter, C. Rolin, S. Verlaak, N. G. Martinelli, D. Beljonne, J. Cornil, J. Genoe, P. Heremans, *Adv. Mater.* **2012**, *24*, 12 1535.

[34] J. Li, G. D’Avino, I. Duchemin, D. Beljonne, X. Blase, *Phys. Rev. B* **2018**, *97* 035108.

[35] G. D’Avino, L. Muccioli, C. Zannoni, D. Beljonne, Z. G. Soos, *J. Chem. Theory Comput.* **2014**, *10* 4959.

[36] H. Mueller, *Phys. Rev.* **1935**, *47* 947 .

[37] J. Colpa, *Physica* **1971**, *56*, 2 185 .

[38] C. K. Purvis, P. L. Taylor, *Phys. Rev. B* **1982**, *26* 4547 .

[39] D. Vanzo, B. J. Topham, Z. G. Soos, *Adv. Funct. Mater.* **2015**, *25*, 13 2004.

[40] K. F. Herzfeld, *Phys. Rev.* **1927**, *29* 701.

[41] E. V. Tsiper, Z. G. Soos, *Phys. Rev. B* **2001**, *64* 195124.

[42] G. D’Avino, D. Vanzo, Z. G. Soos, *J. Chem. Phys.* **2016**, *144*, 3 034702.

[43] B. A. Gregg, *J. Phys. Chem. Lett.* **2011**, *2*, 24 3013.

[44] G. M. Castellan, F. Seitz, *Semiconducting Materials*, Butterworths, London, **1951**.

[45] S. Dhar, A. H. Marshak, *Solid-State Electron.* **1985**, *28*, 8 763.

[46] N. Mott, M. Kaveh, *Adv. Phys.* **1985**, *34* 329.