Synthesis, characterization of Co1-xNixAl2O4 Spinel system and catalytic role in the synthesis of dihydropyrimidinone

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Abstract

In the present investigation, Spinel systems with chemical composition Co$_{1-x}$Ni$_x$Al$_2$O$_4$ ($x = 0.0, 0.25, 0.5,$ and 0.75) have been successfully synthesized by the co-precipitation citrate precursor technique. The phase formation, crystal structure, and impurity check were confirmed by X-ray powdered diffraction (XRD) and Fourier transforms infrared (FTIR) spectroscopy technique. The particle size estimation was done using a transmission electron microscope (TEM). Investigation of magnetic behavior and parameters such as saturation magnetization ($M_S$), coercivity ($H_R$), and retentivity ($M_R$) was done using a vibrating sample magnetometer (VSM). The catalytic activity of prepared spinel systems was explored for the one-pot synthesis of dihydropyrimidinone derivatives. The catalytic product was identified by comparison of melting point and the spectral data (FTIR).

Highlights

The highlights of the research work presented are as follows

- Synthesis of Co$_{1-x}$Ni$_x$Al$_2$O$_4$ via co-precipitation citrate precursor technique
- Particle size distribution was seen to be very narrow.
- Ni$^{+2}$ concentration dependent lattice constant variation was observed.
- Variation in Magnetic properties was seen with increasing Ni$^{+2}$ Concentration
- Increase in percentage yield of dihydropyrimidinone in presence of Co$_{1-x}$Ni$_x$Al$_2$O$_4$

1 Introduction

The nanostructured cobalt aluminate spinels have been the center of attraction due to their unique electrical, magnetic, optical, and catalytic properties. These materials find their application in the field of high-density magnetic recording, microwave devices, applications such as plastics coloring and glass coatings, etc. [1-5,6,7]. The unusual magnetic behavior of FCC structure resulting from geometrical frustration makes it a system of great importance as it is an exemplary system possessing degenerate frustrated ground states which are the result of competition between various exchange interactions and further lead to spin-liquid behavior[8-10]. The spinel composition CoAl$_2$O$_4$ is also known for its catalytic properties arising from high surface area and optimum basicity of the spinel structure and hence have been the subject of an investigation to explore their potential catalytic applications for organic synthesis[11-14]. Studies have shown that these spinel oxides are better catalysts on account of their activity and stability. The phenomenon of synergism in two or three metal-based catalysts is a fascinating aspect of catalytic research [15].

Researchers have synthesized cobalt aluminates using several methods such as microwave combustion method, low-temperature combustion method, sol-gel process, hydrothermal method, reverse microemulsion process, sonochemical process, polymeric-aerosol pyrolysis, freeze-drying, ultrasonic-
assisted-hydrothermal method, chemical vapor deposition (CVD) technique, polymerized complex technique, etc. It has been observed that the properties of resultant cobalt aluminate products are also administered by the method of preparation [16].

In this report the synthesis and characterization of Ni substituted cobalt aluminate nano-powders with compositions CoAl\(_2\)O\(_4\), Co\(_{0.75}\)Ni\(_{0.25}\)Al\(_2\)O\(_4\), Co\(_{0.5}\)Ni\(_{0.5}\)Al\(_2\)O\(_4\), and Co\(_{0.25}\)Ni\(_{0.75}\)Al\(_2\)O\(_4\) using co-precipitation precursor technique using citrate solution has been reported. The report also includes the investigation of the catalytic efficiency effect of Ni\(^{2+}\) substituted cobalt aluminate in synthesizing dihydropyrimidinone via Bignelli reaction.

2. Experimental

2.1 Co-precipitation precursor synthesis

The Spinel structured Co\(_{1-x}\)Ni\(_x\)Al\(_2\)O\(_4\) (x=0.0, 0.25, 0.50, and 0.75) nanocrystalline compositions were synthesized by co-precipitation precursor technique using Citrate solution [17,18]. Stoichiometric quantities of hydrated AR grade reagents Co(NO\(_3\))\(_3\), Ni(NO\(_3\))\(_3\), Al(NO\(_3\))\(_3\) were dissolved in 100mL of double-distilled water in a beaker labeled as ‘A’. In another beaker labeled as ‘B, an equimolar amount of citric acid was dissolved separately in double-distilled water. The citrate solution from beaker B was added dropwise to the solution in beaker A and continuously stirred on a magnetic stirrer for 2 hours with the temperature maintained at 60°C. The temperature was raised further to initiate slow evaporation and reduce the quantity to half, leading to the setting of the desirable gel. The gel thus obtained was subjected to an overnight and slow decomposition process at an elevated temperature of 200°C. The fluffy mass obtained at the end of decomposition was then subjected to heat treatment at 300°C to eliminate the carbonaceous matter. The dried powder was ball milled for 4 hours to attain uniformity in particle size. The processed powders were further heated at finally 800°C in the air for a total time of 10 hours, and were furnace cooled, and stored in airtight containers for characterization. The process of material preparation is as shown in Figure 1.

2.2 Characterization of as-synthesized materials:

The prepared materials were characterized by X-ray powder diffraction technique with Rigaku Miniflex benchtop instrument, using Cu K\(_\alpha\), filtered through Ni absorber. FTIR spectra were recorded on a Shimadzu FTIR instrument (model 8101A) in the range of 4000 - 500 cm\(^{-1}\). The transmission electron micrographs were obtained on a Hitachi 200kV Transmission electron microscope (S.A.I.F., IIT Bombay). The magnetic hysteresis loops for Co\(_{1-x}\)Ni\(_x\)Al\(_2\)O\(_4\) nanocrystalline powders were recorded on Quantum design's 3T vibrating sample magnetometer.

3 Results And Discussion

3.1 X-ray diffraction analysis
The prepared compositions were characterized by X-ray powder diffraction. The X-ray diffraction patterns were recorded at room temperature with a 2θ scanning angle ranging from 20° to 70°.

Figure 2 shows the Rietveld refinement of the XRD pattern for the Co$_{1-x}$Ni$_x$Al$_2$O$_4$ (x = 0.0, 0.25, 0.50, and 0.75) system performed using Fullprof software. The major Bragg reflections were seen at 2θ values of 31.25, 36.14, 38..36, 44.20, 55.92, and 65.51 corresponding to corresponding to the [220], [311], [400], [422], [511] and [440] diffraction planes and were seen to be in accordance with JCPDS data file 44-0160 indicating the formation of pure spinel phase belonging to Fd3m space group without any impurity [19,20]. The goodness of refinement quality is evident from the refinement parameters (c2, R$_{wp}$, R$_{exp}$, R$_{Bragg}$, and RF) listed in table 1.

| Material composition          | c2  | R$_{wp}$ | R$_{exp}$ | R$_{Bragg}$ | RF-FACTOR |
|-------------------------------|-----|----------|----------|-------------|-----------|
| CoAl$_2$O$_4$                 | 1.38| 81       | 68.7     | 57.3        | 39.3      |
| Co$_{0.75}$Ni$_{0.25}$Al$_2$O$_4$ | 1.42| 54.2     | 45.4     | 17.4        | 14.3      |
| Co$_{0.50}$Ni$_{0.50}$Al$_2$O$_4$ | 1.24| 57.5     | 51.7     | 27.9        | 20.6      |
| Co$_{0.25}$Ni$_{0.75}$Al$_2$O$_4$ | 1.59| 83.2     | 65.8     | 63.2        | 44.2      |

The variation of lattice constant ‘a’, x-ray density, and cell volume as a function of Ni$^{+2}$ concentration are shown in Figure 3 (a,b) The lattice constant was seen to decrease initially for x=0.0 to x=0.25 and further showed a marginal rise with increasing Ni$^{+2}$ at the tetrahedral site. The initial decrease in lattice constant can be attributed to smaller ionic radii of Ni$^{+2}$ ions (0.83Å) replacing the larger Co$^{+2}$ ions with ionic radii of 0.88Å [21]. For the compositions beyond x=0.25 i.e. for x=0.50 and 0.75, the increase in lattice constant could be due to electrostatic shielding caused by 3d electrons of Ni$^{+2}$ ions diluting the nuclear dominance and causing marginal expansion in the tetrahedral environment [22]. A trend similar to that of lattice constant variation was observed in the values of cell volume while x-ray density was seen to vary inversely (Figure 2b).

### 3.2 FTIR spectra analysis

Figure 4 shows IR absorption bands for citrate precursor and Co$_{1-x}$Ni$_x$Al$_2$O$_4$ system, in the range of 4000-500 cm$^{-1}$. In general, FTIR spectra of all the prepared spinels show weak absorption bands at about 3450, 1630, and 1385 cm$^{-1}$. The characteristic vibrational frequencies of CoAl$_2$O$_4$ aluminate spinel appeared in the range of 750-400 cm$^{-1}$. The absorption band around 3400 to 3500 cm$^{-1}$ was assigned to stretching vibrations and 160-1400cm$^{-1}$ is assigned to bending vibration respectively which is due to adsorbed moisture. The band observed at 750-500 cm$^{-1}$ can be attributed to the symmetric stretching ($\delta_1$), bending ($\delta_2$), and asymmetric stretching ($\delta_3$) modes of M-O-Al, M-O, and Al-O bonds at tetrahedral and octahedral sites in CoAl$_2$O$_4$ lattice [23]. The broad absorption bands observed at 670 and 560 cm$^{-1}$ are the characteristic vibrational bands representing a typical pattern of CoAl$_2$O$_4$ normal spinel structure, in
agreement with the literature [24]. These bands correspond to the AlO$_6$ units, which is the mainframe of the CoAl$_2$O$_4$ crystal, and ensure the formation of spinel structure. The absorption frequencies observed between 1800-1000 cm$^{-1}$ were assigned to the deformation mode of Al-OH and Co-OH, which is typical of this class of materials [25]. In the Co$_{0.25}$Ni$_{0.75}$Al$_2$O$_4$ nanoparticle spectra (e), the prominent vibrational peaks observed at around 600, 530, and 495 cm$^{-1}$, the absence of a strong absorption band at 560 cm$^{-1}$ and the appearance of a stronger frequency band at around 598 cm$^{-1}$ indicate the substitution of Ni atoms has taken place replacing Co atoms in the lattice. The first two absorption peaks are attributed to the intrinsic stretching vibrations of the M-O at tetrahedral sites, while the lower mode is assigned to the stretching vibration of the M-O at the octahedral site[23-27]. No other impurity or organic matter absorption bands were observed indicating the purity of the materials, in agreement with the results obtained by XRD. FTIR measurements were also helpful to illustrate the formation of spinel from the precursor.

3.3 VSM data analysis

The magnetic hysteresis loops obtained at room temperature for Co$_{1-x}$Ni$_x$Al$_2$O$_4$ nano-powders are shown in Figure 5

The samples were seen to exhibit paramagnetic behavior [1]. The values of saturation magnetization ($M_S$), coercive field ($H_C$), and remnant magnetization ($M_R$) are listed in Table 3.

| Sample                  | Saturation Magnetization ($M_S$) emu/g | Coercive field ($H_C$) Oe | Remnant Magnetization ($H_R$) emu/g | Squareness | Magnetic Moment $m$ (Bohr Magneton) |
|-------------------------|---------------------------------------|----------------------------|-------------------------------------|------------|-----------------------------------|
| CoAl$_2$O$_4$            | 0.56                                  | 16.04                      | 0.05                                | 0.083      | 0.018                             |
| Co$_{0.75}$Ni$_{0.25}$Al$_2$O$_4$ | 0.51                                  | 24.06                      | 0.03                                | 0.058      | 0.016                             |
| Co$_{0.50}$Ni$_{0.50}$Al$_2$O$_4$ | 0.72                                  | 72.18                      | 0.04                                | 0.061      | 0.022                             |
| Co$_{0.25}$Ni$_{0.75}$Al$_2$O$_4$ | 0.66                                  | 24.34                      | 0.04                                | 0.051      | 0.020                             |

The $M_S$ values were seen to vary with Ni$^{+2}$ content and showed a maximum value of 0.72emu/g for x=0.5 and a minimum of 0.51emu/g for x=0.25 while the remnant magnetization was seen to remain almost constant with values ranging between 0.03emu/g to 0.05emu/g. The squareness values calculated using the following equation (1) were found to be very low confirming the paramagnetic behavior of the nano-powders. However, these values were much higher than the reported squareness limit for superparamagnetic materials [28-30]. The magnetic moment $m$ calculated for all the samples using equation (2) was seen to remain in the range of 0.016 to 0.022 Bohr magneton with increasing Ni concentration in the spinel lattice [31-35].
Where \( M_x \) is the molecular weight of the composition.

The anisotropy constant was calculated using equation 3 given below showed a trend similar to that of HC. The plot of variation in anisotropy constant ‘\( K \)’ with increasing Ni concentration is shown in figure 6.

\[
K = \frac{M_S H_C}{0.96}
\]  
Equation 3

### 3.4 TEM image analysis

The transmission electron micrographs obtained on CoAl\(_2\)O\(_4\) and Co\(_{0.25}\)Ni\(_{0.75}\)Al\(_2\)O\(_4\) nanocrystalline powders along with particle size distribution histograms are shown in Figure 7. The observed particle size of these nanocrystalline spinel aluminates was in the range of 20 nm to 47 nm.

### 3.5 The Bignelli reaction as a model catalytic application

The as-synthesized spinel Co-Ni aluminate compositions were explored for catalytic efficiency of synthesizing dihydroxymidinone derivative as a model test reaction as shown in Figure 8. A solution of Benzaldehyde (10 mmol, 1.06 g), ethylacetoacetate (13mmol, 1.69 g), and urea (15 mmol, 0.90g) was refluxed at 85-90°C in ethanol in the presence of materials (0.2g) under investigation, for 3 hours. On completion of the reaction, the catalyst was filtered off from the mixture, and the filtrate was collected in crushed ice. The product obtained was recrystallized using ethyl acetate. The synthesized product was identified by comparison of melting point (mp) and the spectral data (FTIR).

The catalytic product was confirmed by melting point (mp) and FTIR spectroscopy. Table 4 summarises the yield obtained (%) for the dihydroxymidinone product. The results obtained using Ni substituted cobalt ferrite nanopowders as catalyst are comparable to those obtained by Ezzat Rafiee et. al. in which dihydroxymidinone was reported to synthesize using heteropoly acids such as H\(_3\)PW\(_{12}\)O\(_{40}\), H\(_3\)PMo\(_{12}\)O\(_{40}\) [35]. The efficiency nanopowders of Co\(_{1-X}\)Ni\(_X\)Al\(_2\)O\(_4\) can be enhanced to produce higher yield by increasing the Ni concentration in the composition as these nanopowders possess higher surface area in comparison to that of heteropoly acids.

Table 4 Yield of dihydroxymidinone and their melting points.
Nanocatalysts | % Yield | Physical constant (mp)
--- | --- | ---
In the absence of a catalyst | 30% | 207°C
CoAl₂O₄ | 45.91% | 205°C
Co₀.₇₅Ni₀.₂₅Al₂O₄ | 46.34% | 205°C
Co₀.₅₀Ni₀.₅₀Al₂O₄ | 49.79% | 207°C
Co₀.₂₅Ni₀.₇₅Al₂O₄ | 51.93% | 206°C

Figure 9 shows the FTIR spectra of the product and the characteristic absorbance peaks which are summarised in Table 5, confirming the formation and purity of dihydropyrimidinone. The melting point of the purified product was found to be ranging between 205°C to 207°C which again indicated the purity of the reaction product [36-40].

Table 5 Characteristic absorption peaks of dihydropyrimidinone.

| Observed absorption bands (cm⁻¹) | Type of stretching |
|---|---|
| 3337.39 | N-H stretch |
| 3115.53 | N-H stretch |
| 2979.46 | C-H stretch |
| 1725.29 | C=O ester stretch |
| 1702.88 | C=O amide stretch |
| 1650.29 | C=C stretch |
| 1089.21 | Monosubstituted aromatic ring |

As the atomic concentration of Al remains practically the same in all the Co₁₋ₓNiₓAl₂O₄ spinel nanomaterials, it may have an indirect role in the activity. The observed increase of 30 to 51% in the yield percentage with the Ni substitution and its atomic concentration in CoAl₂O₄ crystal structure, may be attributed due to atomic size variation and synergistic effect of both Co⁺² and Ni⁺² ions.

4 Conclusion

Nanocrystalline powders of Ni⁺² substituted Cobalt aluminate with chemical composition Co₁₋ₓNiₓAl₂O₄ (x=0.0, 0.25, 0.5, 0.75) were successfully prepared by co-precipitation precursor technique using citrate solution. Samples were seen to exhibit pure spinel phase and particle size ranging between 25 nm to 40 nm. Structural parameters such as lattice constant, cell volume, and X-ray density were seen to change with varying Ni⁺² concentrations.

The FTIR spectra confirmed the existence of characteristic spinel bands confirming the phase purity. All the samples exhibit paramagnetic behavior with saturation magnetization varying between 0.7 to 0.5 emu/g. The application of Co₁₋ₓNiₓAl₂O₄ nanopowders as a catalyst in the synthesis of dihydropyrimidinone was investigated. The reaction showed an enhancement of 15 percent in percentage yield with pure CoAl₂O₄ employed as catalyst. The percentage yield was seen to increase further by 21
percent with increasing Ni content in the spinel structure. The efficiency of Co$_{1-x}$Ni$_x$Al$_2$O$_4$ can be improved further by increasing Ni concentration and can be used as more efficient catalyst due to its larger surface area.

**Declarations**

We the authors of the manuscript entitled “Synthesis, characterization of Co$_{1-x}$Ni$_x$Al$_2$O$_4$ Spinel system and catalytic applications in the synthesis of dihydropyrimidinone” declare that

1. The article is original.
2. The article has been written by the stated authors who are all aware of its content and approve its submission.
3. The article has not been published previously
4. The article is not under consideration for publication elsewhere
5. No conflict of interest exists, or if such conflict exists, the exact nature must be declared.

If accepted, the article will not be published elsewhere in the same form, in any language, without the written content of the publisher.

The present work entitled Synthesis, characterization of Co$_{1-x}$Ni$_x$Al$_2$O$_4$ Spinel system and catalytic applications in the synthesis of dihydropyrimidinone includes the synthesis and characterization of Ni substituted cobalt aluminate nano-powders with compositions CoAl$_2$O$_4$, Co$_{0.75}$Ni$_{0.25}$Al$_2$O$_4$, Co$_{0.5}$Ni$_{0.5}$Al$_2$O$_4$, and Co$_{0.25}$Ni$_{0.75}$Al$_2$O$_4$ using co-precipitation precursor technique using citrate solution. The report also includes the investigation of the catalytic efficiency effect of Ni$^{2+}$ substituted cobalt aluminate in synthesizing dihydropyrimidinone via Bignelli reaction.

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Figures
Figure 1

Co-precipitation precursor technique using Citrate solution
Figure 2

Rietveld refinement of XRD patterns obtained for Co1-xNixAl2O4
**Figure 3**

Variation of (a) Lattice constant and (b) X-ray density, cell volume with Ni$^{+2}$ concentration

**Figure 4**

FTIR spectra of (a) Cobalt Nickel Citrate Precursor (b) CoAl$_2$O$_4$ (c) Co0.75Ni0.25Al$_2$O$_4$ (d) Co0.5Ni0.5Al$_2$O$_4$, and (e) Co0.25Ni0.75Al$_2$O$_4$. 
Figure 5

Hysteresis loops obtained for Co1-XNiXAl2O4 nanoparticles
Figure 6

Variation of anisotropy constant ‘K’ with Ni concentration
Figure 7

Transmission electron micrographs and particle size distribution histograms of CoAl2O4 and Co0.25Ni0.75Al2O4

\[
\begin{align*}
\text{Catalyst} & \quad \text{Heat} \\
\text{Dihydropyrimidinone} 
\end{align*}
\]

Figure 8

Bignelli reaction as a model catalytic application
Figure 9

FTIR spectra of dihydropyrimidinone using Co1-XNiAl2O4 nanocatalysts.

Supplementary Files

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