Fabrication of NiO/NiCo$_2$O$_4$ Mixtures as Excellent Microwave Absorbers

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Abstract

The NiO/NiCo$_2$O$_4$ mixtures with unique yolk-shell structure were synthesized by a simple hydrothermal route and subsequent thermal treatment. The elemental distribution, composition, and microstructure of the samples were characterized by transmission electron microscopy (TEM), X-ray diffraction (XRD), and scanning electron microscope (SEM), respectively. The microwave absorption property was investigated by using vector network analysis (VNA). The results indicated that the excellent electromagnetic wave absorption property of the NiO/NiCo$_2$O$_4$ mixtures was achieved due to the unique yolk-shell structure. In detail, the maximum reflection loss (RL) value of the sample reached up to $-37.0$ dB at 12.2 GHz and the absorption bandwidth with RL below $-10$ dB was 4.0 GHz with a 2.0-mm-thick absorber. In addition, the NiO/NiCo$_2$O$_4$ mixtures prepared at high temperature, exhibited excellent thermal stability. Possible mechanisms were investigated for improving the microwave absorption properties of the samples.

Keywords: NiO/NiCo$_2$O$_4$, Mixtures, Yolk-shell, Electromagnetic wave absorption

Background

Recently, with arising and development of wireless communication and the wide application of electronic devices, electromagnetic contamination has become serious problems to the electronic equipment [1]. High-power electromagnetic wave in regional environment can interfere with each other, which can result in the damage of communication system or even cause serious accidents, such as missile error, aircraft crash, and other disastrous consequences. Therefore, to develop a high-efficiency electromagnetic wave absorption (EMW) absorber with strong absorption, wide bandwidth, small thickness, and lightweight is highly desirable.

Currently, the study on EMW absorbers is mainly concentrated on transition-metal oxides [2, 3], binary metal oxides [4], carbonaceous materials [5–7], conducting polymers [8], magnetic materials [9–12], Metal–organic-frameworks materials [13, 14], and graphene-based hybrid materials [15–21]. At present, NiO and NiCo$_2$O$_4$ have attracted tremendous interest due to its unique properties in the absorption intensity and frequency bandwidth of electromagnetic wave. As we all know, NiCo$_2$O$_4$ is a hybrid transition metal oxide with excellent electrical and electrochemical properties [22, 23]. The potential of NiCo$_2$O$_4$ [24] and NiCo$_2$O$_4$@PVDF composite [25] for electromagnetic wave absorption has been studied. Interestingly, recent research also demonstrated the potential of NiO and its related mixtures in the application of microwave absorption [26, 27]. Therefore, the combination of NiO and NiCo$_2$O$_4$ for the preparation of electromagnetic wave absorbing materials has become a new research field. For instance, Liu et al. [28] conducted some explorations on the EMW absorption properties of NiCo$_2$O$_4$/Co$_3$O$_4$/NiO composites. Their results demonstrated that the sample exhibited a maximum RL value of $-28.6$ dB at 14.96 GHz.

Porous NiO/NiCo$_2$O$_4$ lotus root-like nanoflakes were demonstrated by Liang and co-workers as a promising candidate for a microwave absorbent [29]. The NiO/NiCo$_2$O$_4$ (60 wt%)-wax hybrid showed the strongest EMW absorption with the RL value of $-47$ dB at
13.4 GHz. However, the method used in the preparation of the NiO/NiCo$_2$O$_4$ hybrid process is too complicated to be suitable for mass production. As a result, the development of a facile method for the preparation yolk-shell NiO/NiCo$_2$O$_4$ mixtures with excellent EMW performance is still an intriguing topic.

Herein, we report a simple hydrothermal method and subsequent post-thermal treatment to prepare the NiO/NiCo$_2$O$_4$ mixtures with unique yolk-shell structure. Results indicated that the obtained samples exhibited excellent microwave absorbing performance. The relationship between structure, surface morphologies, and the microwave absorbing performance was also discussed. The current study will greatly expand the application scenario of the NiO/NiCo$_2$O$_4$ mixtures as electromagnetic wave absorber.

**Methods**
The precursor was first prepared using a simple hydrothermal method. In a typical synthesis, 1 mmol of Ni(NO$_3$)$_2$, 2 mmol of Co(NO$_3$)$_2$·6H$_2$O, and 0.6 mol of urea (H$_2$NCONH$_2$) were dissolved in 5 mL isopropanol (C$_3$H$_8$O) and 25 mL of deionized water and then stirred for 0.5 h to make them full dispersed. Then, the resultant solution was transferred to a polytetrafluoroethylene reactor and reacted at 120 °C for 12 h. After that, the autoclave was cooled down to room temperature naturally. Then, the sample was collected by centrifuge and washed several times with alcohol and deionized water, respectively. The obtained wet powder was dried at 60 °C for 10 h in a vacuum oven. Pink precipitates were further calcined at 350 °C, 450 °C, 550 °C, and 650 °C for 3.5 h under atmospheric conditions, respectively. The reagents used in the assay were all analytically pure and used without further purification.

The crystalline phases of the calcined products were characterized by X-ray diffractometer (XRD, PANalytical, Empyrean) using Cu Kα radiation (λ = 1.54178 Å, 40.0 kV). Structures, morphologies, composition, and elemental distribution of the samples were observed by using scanning electron microscopy (SEM, JEOL-6610LV) and transmission electron microscopy (TEM, JEM-2100, INCA-X-Max80). The electromagnetic parameters of the obtained samples were examined by the vector network analyzer (VNA, AV3629D) using transmission-reflection mode in the frequency range of 2.0–18.0 GHz at room temperature. The samples with different annealing temperatures (350 °C, 450 °C, 550 °C, 650 °C) are labeled as S1, S2, S3, and S4, respectively to the convenience of this description.

**Results and Discussion**
The X-ray characteristic spectra of the samples at different annealing temperatures are shown in Fig. 1. Comparing the standard cards of NiO (PDF#44-1159) and NiCo$_2$O$_4$ (PDF#20-0781), it is found the diffraction peaks of the samples with annealing temperature of 650 °C and 550 °C correspond to NiO (2θ = 37.2°, 43.3°, and 62.9°) and NiCo$_2$O$_4$ (2θ = 31.1°, 36.7°, 44.6°, 59.1°, and 64.9°), respectively. The XRD patterns show that the NiO/NiCo$_2$O$_4$ mixtures are successfully synthesized by using the raw materials mentioned in the experiment. However, the diffraction peaks of NiO are not found in samples with annealing temperature less than 550 °C, indicating that high temperature is favorable for the formation of NiO. In the hydrothermal reaction stage, due to the participation of urea, we obtain a small amount of NiCO$_3$, which can be decomposed to NiO and CO$_2$ at high temperature. At the same time, with the increase of annealing temperature, the crystallinity of NiCo$_2$O$_4$ crystals is also optimized, which means that the samples can be used in a high-temperature environment.

SEM images of all samples are shown in Fig. 2. As shown in the micrographs of the sample, most of the samples exhibited microspheres of different diameters with a vast of radial nanowires on the surface. However, with the annealing temperature increasing, the surface of the sample cracks and a mass of pores are generated, such as the sample corresponding to the annealing temperature of 650 °C.

In order to further investigate the microstructure and the distribution of NiO and NiCo$_2$O$_4$ in the NiO/NiCo$_2$O$_4$ mixtures, transmission electron microscopy (TEM), and electron diffraction spectrum (EDS) was used to measure the sample with an annealing temperature of 650 °C. From Fig. 3a, b, we can see a typical yolk-shell structure. In Fig. 3c, Co elements are mainly concentrated on the kernel part. Therefore, it can be inferred that NiCo$_2$O$_4$ is mainly distributed within the nucleus. According to the distribution of Ni elements shown in Fig. 3d, there is a clear gap between the shell layer and the kernel part, which is somewhat different from the distribution of Co elements. Combining with the XRD patterns of the sample, it can be inferred that NiO is mainly distributed on the outer sphere of the whole hollow core-shell structure. The composition is verified by EDS spectroscopy as shown in Fig. 3d [30]. In addition, the Cu, Cr, and C elements shown in the EDS spectrum belong to the measuring instrument itself.

Furthermore, the porous structure can reduce the effective permittivity of the material, which is advantageous to impedance matching [31, 32]. According to
the subsequent analysis of the measured electromagnetic parameters of the samples, it is believed that this change may be beneficial to improve the degree of impedance matching of the samples, and then improve the electromagnetic wave absorption effect.

It is well known that the electromagnetic parameters of a material, including relative permittivity \((\varepsilon_\gamma=\varepsilon'-j\varepsilon'')\) and relative complex permeability \((\mu_\gamma=\mu'-j\mu'')\), play an extremely important role in the EMW absorption performance. The real part of the complex permittivity \((\varepsilon')\) and the complex permeability \((\mu')\) show the storage capacity of the absorbing material for electrical and magnetic energy, while the imaginary part show loss capacity of electrical and magnetic energy [33, 34]. When these two electromagnetic parameters are close, this means that the material has a good impedance match. In this experiment, the electromagnetic parameters of samples were measured by dispersing the composites in a paraffin matrix with loading of 30wt% in the frequency range of 2–18 GHz. By substituting the measured electromagnetic parameters into the following formula, the reflection loss ability of the sample for an electromagnetic wave at different thicknesses can be simulated and calculated [35].

\[
Z_{in} = Z_0 \sqrt{\frac{\mu_\gamma}{\varepsilon_\gamma}} \tanh\left(\frac{2\pi f d}{c} \sqrt{\mu_\gamma \varepsilon_\gamma}\right) \quad (1)
\]

\[
RL(\text{dB}) = 20 \log\left|\frac{Z_{in} - Z_0}{Z_{in} + Z_0}\right| \quad (2)
\]

Where \(\varepsilon',\varepsilon'',\mu',\) and \(\mu''\) represent the real and imaginary parts of permittivity and permeability, respectively. The \(f\) value is the frequency of electromagnetic wave, \(d\) is the thickness of the absorber, \(Z_0\) is the impedance of free space, \(Z_{in}\) is the normalized input impedance, and \(c\) is the velocity of light in free space [36].

According to Formula (1)–(2), it can be concluded that when the reflection loss reaches \(-20\) dB, the corresponding material absorbs about 99% of the EMW, which means that the sample can be applied to actual needs [37].

The real part \((\varepsilon')\) and imaginary part \((\varepsilon'')\) of the permittivity of the samples are shown in Fig. 4a, b, respectively, and the changes in the real and imaginary parts of the permittivity of the sample at different temperatures are carefully compared. It is shown that
the value of $\varepsilon'$ decreases from 72.6 to 30.3 with increasing frequency for the sample with the annealing temperature of 350 °C. However, the $\varepsilon''$ value of the sample is showing different trends and the overall trend of decreasing in the test frequency range. There is a large fluctuation in the range of 7.1–10.4 GHz, which is mainly caused by dielectric relaxation. Obviously, the $\varepsilon'$ and $\varepsilon''$ values of NiO/NiCo$_2$O$_4$ mixtures (550 °C and 650 °C) do not change significantly, compared with NiCo$_2$O$_4$ particles. It can be clearly seen from Fig. 3, the $\varepsilon'$ of the composites decreases as the annealed temperature increased. The electromagnetic parameters of the S3 and S4 have very similar trends and are different from the S1 and S2. In the frequency range of test, $\varepsilon'$ and $\varepsilon''$ of S3 and S4 varied in the range of 15.3 to 8.5 and 4.1 to 2.0, respectively. Based on the theory of free electrons, the high $\varepsilon''$ value of samples resulted in the high conductivity [38]. However, too high conductivity leads to the mismatching between permittivity and permeability, which is not favorable to the microwave absorption performance. When NiO crystal with higher electric resistivity is combined with NiCo$_2$O$_4$, the formation of electric conducting networks of the NiCo$_2$O$_4$ is prevented, thereby reducing the conductivity of the composites. For all samples, the $\mu'$ and $\mu''$ of complex permeability in the whole frequency range, are very close to 1 and 0 even to negative, respectively [39, 40], (Additional file 1; Figure S1) which implied that the magnetism of samples is small and negligible.

In general, the reflection loss for electromagnetic wave material is related to the dielectric loss factor ($\tan\delta_e = \varepsilon''/\varepsilon'$). As shown in Fig. 4c, the dielectric loss factors for S3 and S4 are significantly smaller than for S1 and S2. The maximum dielectric loss factors for S3 and S4 are 0.69 (10.9 Hz) and 0.57 (18 Hz), respectively. The impedance matching ratio is widely used to demonstrate the dielectric loss ability of microwave absorbers [41]. The impedance matching ratio of the samples can be denoted as Eq. (3).

$$Z_t = \left| \frac{Z_m}{Z_0} \right| = \sqrt{\mu'/\varepsilon'} \tanh \left( j(2\pi f d/c) \sqrt{\mu'/\varepsilon'} \right)$$  \hspace{1cm} (3)$$

In the Eq. (3), $f$, $c$, $Z_m$, $Z_0$, and $Z_t$ is the attenuation constant, frequency, velocity of light, the input impedance of absorber, the impedance of free space, and the impedance matching ratio value, respectively. In order to further illustrate the electromagnetic loss properties of the samples, the impedance matching ratio of the material is introduced and shown in Fig. 4d. Interestingly, we found that the impedance matching ratio of the NiO/NiCo$_2$O$_4$ mixtures is significantly higher than that of the S1 and S2. As a result, the former is more effective in absorbing electromagnetic wave.

It is obvious that the RL curve of samples can be used to reflect their microwave absorption performance. Based on the transmission line theory, it is possible to simulate and calculate the microwave absorbing parameters in the thickness range of 1.0–
5.0 mm, according to the electromagnetic parameters. The theoretical RL curves of the samples calcinated at different temperatures in the frequency range of 2–18 GHz is shown in Fig. 5. It is generally considered that when the RL is lower than −10 dB, the absorption rate of the electromagnetic wave of the sample can reach more than 90% [42], which is a typical performance index to be achieved by the application of the microwave absorption material. According to Fig. 5a and b, it is clearly indicated that the RL values of S1 and S2 are relatively poor and there is no bandwidth under −10 dB. However, with the increase of NiO crystallinity in the samples, the minimum reflection loss of the NiO/NiCo$_2$O$_4$ mixtures is much lower than −10 dB. Such as S4 shown in Fig. 5d, the range of frequency below −10 dB correspond to the value of RL is 10.6~14.6 GHz and the bandwidth is 4.0 GHz. Meanwhile, we find that the minimum reflection loss reaches to −37.0 dB at 12.2 GHz with the absorber thickness of 2.0 mm. In order to reflect the relationship between the RL and the thickness of the sample intuitively, the corresponding 3D contour curves are shown in Fig. 5. From the simulated electromagnetic wave reflection loss image, S4
would show excellent wave absorbing performance in the thickness range of 1.5–5.0 mm.

In addition to the inherent dipoles in NiCo$_2$O$_4$ and NiO phases, the defect dipoles are also generated due to the formation of lattice defects caused by phase transformation [28]. As a result, these dipoles would produce dielectric loss by orientation polarization relaxation in alternating electromagnetic fields. Interestingly, interfacial polarization relaxation will occur in NiO/NiCo$_2$O$_4$ mixtures with many heterogeneous interfaces, resulting in an enhanced dielectric loss. As shown in XRD pattern, when the annealing temperature reaches to 550 °C, some characteristic peaks such as 37.2°, 43.3°, and 62.9° can be found, which demonstrates the generation of NiO. The intensity of the diffraction peak of NiO at 35.49° is reinforced following the temperature, implying that more NiO crystals were produced.

In order to intuitionistic illustrate the possible mechanism, the diagrammatic map named Fig. 6 was provided. According to the figure, NiO/NiCo$_2$O$_4$ mixtures exhibit prominent microwave absorption property, which may be the following reasons. First, the NiO/NiCo$_2$O$_4$ mixtures have a rich heterogeneous interface, resulting in strong interfacial polarization relaxation, which leads to large dielectric losses. Secondly, the void space and interspaces in the shell-core structures enable the full exposure of NiO/NiCo$_2$O$_4$ mixture materials to the atmosphere, which facilitates the introduction of electromagnetic waves and produces dielectric resonance [43, 44]. Thirdly, the unique yolk-shell structure of NiO/NiCo$_2$O$_4$ mixtures can reflect and absorb the absorbed electromagnetic waves.
Fig. 5 EM reflection loss curves of the samples. Where a–d represent the reflection loss curves of the samples with the annealing temperatures of 350 °C, 450 °C, 550 °C and 650 °C, and the images on the right correspond to the 3D reflection losses of the samples, respectively.
waves multiple times to enhance the loss of electromagnetic waves in the sample [45, 46].

Conclusions
The NiO/NiCo$_2$O$_4$ mixtures with yolk-shell structure were prepared by hydrothermal method and followed by annealing at high temperature. When the annealing temperature is 650 °C, the NiO/NiCo$_2$O$_4$ mixtures exhibit the best microwave absorption properties, which is much better than the performance of pure NiCo$_2$O$_4$ and similar composites. The enhanced microwave absorption ability of the composites is mainly attributed to the interfacial polarization relaxation, orientation polarization relaxation caused by defect dipoles, and the unique yolk-shell structure. It is believed that such composites will be promising for widespread applications in the microwave absorption field.

Additional file

Additional file 1: Figure S1. (a) Real part and (b) imaginary part of the complex permeability of the samples. (EPS 2207 kb)

Abbreviations
EMW: Electromagnetic wave absorption; RL: Reflection loss

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Availability of Data and Materials
The datasets and supporting information supporting obtained in this paper are included in this article.

Authors’ Contributions
ZL conceived the project and supervised the study. XC performed the experiment and accomplished the whole manuscript. XZ and SP did the tests. ZL and QL guide the theoretical analysis for the results of the tests. QL revised the manuscript. All authors read and approved the final manuscript.

Competing Interests
The authors declare that they have no competing interests.

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