Blended additive manipulated morphology and crystallinity transformation toward high performance perovskite solar cells†

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The efficiency of a perovskite solar cell is highly affected by the crystal quality and morphology of its perovskite layer. A facile strategy of using blended organic additives to regulate the film’s crystallinity and coverage is studied in this work. The coordination behavior and sublimation temperature of the additives are the essential characteristics which determine the quality of the perovskite films. By introducing a certain amount of MAI, a kind of crystallinity-preferred additive, to partially replace MAAc, a coverage-preferred additive, the film’s crystallinity has been obviously improved at no expense of coverage, which results in a lower trap density, higher efficiency and better stability.

Introduction

Organic–inorganic metal halide perovskites (OIMHPs) have drawn extensive attention due to their rapidly increasing power conversion efficiency (PCE) to over 22% within only 7 years.1–3 Except for the superior optoelectronic properties of the perovskite layer, the feasibility of the solution process also makes them promising for both laboratory studies and industrial production.4–6 In ideal perovskite solar cell (PSCs) devices, the perovskite layer should be both highly uniform and with high crystallinity to reduce energy loss during carrier diffusion and extraction.7–9 A stoichiometric precursor solution which contains equal molar amounts of lead iodine (PbI2) and methylammonium iodide (CH3NH3I3 or MAI) dissolved in an organic solvent such as N,N-dimethylformamide (DMF), butyrolactone (GBL) or dimethylsulfoxide (DMSO) is the most typical precursor solution prescription. Because of the strong coordination bonds between solvent (SV) and PbI2, the perovskite precursor presents its composition mostly in the form of SV–PbI2–MAI, SV–PbI2–SV and MAI–PbI2–MAI complexes in solution.10–11 The retarded nucleation rate and heterogeneous growth nature of the coordination complexes during the evaporation of non-coordinated solvent eventually lead to the formation of bulk rod-like morphologies. As the following removal of the residual coordinated solvent molecules is a solid state decomposition reaction, it still maintains very poor film morphology.12

Taking these into consideration, to replace the SV molecules coordination by introduction of stoichiometric ratio of additive is proved to be an effective way to enhance perovskite crystal quality. The recipes of PbI2 : MACI (1 : 3), PbI2 : MAI : MACI (1 : 1 : 1), and PbI2 : PbCl2 : MAI (1 : 1 : 4) recipes have developed to synthesis the MAPbI3 , Clx perovskite films.14–17 Other precursor solutions, such as employing MAAc directly as additive in the precursor solution, could deliver very smooth and compact films, whereas the crystal size is much smaller, normally resulting in reduced photoluminescence intensity and lifetime.18,19 Besides, other solvent additives, such as 1-chloronaphthalene, also can enhance the film uniformity via interacting with Pb2+ to form a chelate complex.20 In many single additive related cases, either the film uniformity or the crystallinity could be obviously improved in comparison with non-additive PbI2 : MAI (1 : 1) system. Generally, there is an offset of the film uniformity and crystallinity because they are conflicting as the high uniform film acquires a fast nucleation process, which normally reduces the time for grain growth and produces large amount of grain boundary defects in the films.

Herein, we employ a strategy of blended-additives engineering to regulate perovskite film morphology and crystallinity, in which MAAc ensures film smoothness and coverage, and MACI facilitates perovskite film crystallinity. The component of the precursor in our method is MAPbI3 with equal molar of blended xMAAc−(1 − x)MACI additives. By appropriate change the ratio of these two components, perovskite film crystallinity and morphology will be regulated accordingly.

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Benefiting from the modified film properties, the optimized device in this work shows lower trap density, improved stability and better efficiency.

Result and discussion

Fig. 1A–C shows the typical morphology of MAPbI$_3$ films using MAAc, MACl and blended additives respectively. As can be seen in Fig. 1A, MAPbI$_3$ film with MAAc additive (MAPI-MAAc) shows very smooth and compact film morphology, in which with small crystals of about 150 nm. As for the film with MACl additive (MAPI-MACl), the size of perovskite crystals distinctly enhanced to about 600 nm, but with apparent pinholes between them (Fig. 1B). After replacing certain amount of MACl by MAAc, the fabricated sample shows enhanced film crystallinity at no expense of film coverage as shown in Fig. 1C. The feasibility of this regulating strategy is evidenced in Fig. S1† by a series of blended organic additives (BOAs) films. When MACl component increases (MAAc amount decreases), the perovskite crystal size correspondingly shows an increases tendency. However, excessive amount of MACl in the component will result in the shrink of the perovskite crystals which destroys the film coverage. When $x$ reaches to 0.8, the fabricated films exhibit optimized properties in the consideration of both film coverage and crystal size, which can be seen in Fig. 1C. Fig. 1D shows the UV-vis absorption spectra of the three films. All the three films show very similar absorption lines which indicate their similar optical properties. The intensity of MAPI-BOAs is slightly stronger compared to the other two films in the region below 700 nm, which is mainly attributed to improvement of the uniformity compared with the MAPI-MACl film. The X-ray diffraction (XRD) patterns for these three films are also measured to compare their phase composition and crystal property. As can be seen in Fig. 1E, all these three films show characteristic diffraction peaks of $2\theta = 14.0^\circ$ and $28.4^\circ$, which indicate their complete conversion to MAPbI$_3$ perovskite from precursor solutions. The peaks intensity of MAPI-MACl and MAPI-BOAs films however are much stronger than that of MAPI-MAAc film, which indicates their preferential crystal growth along (110) faces and higher crystallinity. From what have been shown above, the MAPI-BOAs film with 80% MAAc and 20% MACl still retains full-coverage morphology but with obviously improved crystallinity.

Numerous additive-related works have been done before, but the properties of the fabricated films differ as the additives vary in these works and the mechanism is still not very clear.$^{21,22}$ It’s been known that the crystallization process can be divided into two simultaneous processes, which are nucleation and crystal growth.$^{23}$ The distinct morphology and crystal quality of these three films should be ascribed to their very different crystallization process. We hereafter investigated their crystallization process from the stage of precursor solutions to final perovskite films, which should be of great importance to the understanding of their morphology distinction. Perovskite precursor solutions, due to their strong coordination effects between molecules, are deemed as colloidal solutions which contain colloidal property. While for the other three solutions with different additives and pristine MAPbI$_3$ precursor solution, to evaluate their colloidal particle size distribution. In Fig. 2A of pristine MAPbI$_3$ precursor solution, the particle size distribution peak is at around 512 nm, which indicates its well-defined colloidal property. While for the other three solutions with additives, the particle sizes decrease to varying extents. For MAPI-MACl sample, the particle sizes is about 412 nm [Fig. 2B], and after the blending of MAAc, the size decrease to 376 and 236 nm respectively for MAPI-BOAs

Fig. 1  (A–C) SEM images of MAPI-MAAc, MAPI-BOAs and MAPI-MACl perovskite films. (D and E) The corresponding UV-vis absorption spectra and XRD patterns of the three samples.
process of MAPbI$_3$ from the precursor solutions with/without additives. The samples are prepared on TiO$_2$ scaffolds by blading and low temperature annealing treatment is employed to prolong this process for observation. Although the testing condition is different from the normal treating process of perovskite films, the visible nucleus and growth processes are helpful for us to understand the additives’ functions. At the origin stage, the precursor solutions are spread evenly on the substrates. At the first stage of 30 seconds, all the precursors with/without additives start to nucleate but with different nucleation density. For MAPI-MACI film, very few nucleation centers are formed on the surface. The number of nucleation sites grows when the additive changes to BOAs. For MAPI-MAAc film, the nucleation density grows prominently, and these crystal nuclei scatters averagely on the glass, but the size of them decrease compared to that of the former two. The small size distribution of the intermediate phase in mother solution should contribute to the size contraction of the crystal nucleus precipitating on substrate and this is one aspect from where, the small size of MAPI-MAAc perovskite crystals forms. For MAPI film, very dense nuclei are formed on the substrate, and rod-like crystals seems to appear at this very early stage. At about 1 minute, more nucleuses are grown on the substrate for these films, which highly follow the characteristics of their former stages. Besides, the crystal size grows larger for MAPI-MACI and MAPI-BOAs samples, but not obvious for MAPI-MAAc and MAPI films. When it comes to 5 minutes, the nucleuses continue to grow bigger only for MAPI-MACI and MAPI-BOAs films. At the last stage of 10 minutes, sparse and large crystals are grown on the surface of MAPI-MACI film, while for MAPI-MAAc film, very compact and small crystals are arranged. For MAPI-BOAs film, the crystal density is larger than that of MAPI-MACI film, and the crystal size shows no obvious reduction. However, for MAPI film, the film morphology shows no change at this stage. The difference in nucleation density and crystal size of these films at the same stage is associated with the properties of their different intermediates.

As can be concluded from what have been discussed above, the coverage and crystallinity of the films by MACI blending is enhanced. To further assess the film quality by MACI blending, a series of optoelectronic properties are measured. Fig. 3A shows the time-resolved photoluminescence (TRPL) decay measurements of MAPI-MAAc and MAPI-BOAs films (on glasses). The curves can be well fitted with a biexponential function containing both slow and long lifetimes, which are $\tau_1$ and $\tau_2$ respectively. We assign $\tau_1$ as the decay which is closely
related to the non-radiative recombination by the traps on film surface and grain boundaries, and \( \tau_2 \) as the decay of radiative recombination from bulk perovskite which is associated with bulk carrier recombination.\(^\text{16}\) For MAPI-MAAc sample, \( \tau_1 \) is calculated to be 2.5 ns. While for MAPI-BOAs sample \( \tau_1 \) increases to 4.3 ns, which indicates its fewer non-radiative recombination on the film surface and between grain boundaries. MAPI-BOAs film shows longer \( \tau_2 \) of 18.9 ns compared to 15.0 ns of MAPI-MAAc sample, indicating the recombination in the bulk film is also efficiently reduced. For comparison, the TRPL decay of MAPb-MACl film was also measured as shown in Fig. S5A.\(^\text{1}\) The calculated \( \tau_1 \) and \( \tau_2 \) are 5.6 ns and 34.1 ns, respectively. It was indicative of the positive effect of the MACl additive on the film crystallinity. Fig. 3B is the normalized PL spectra of MAPI-MAAc and MAPI-BOAs films. Both samples show very similar PL peaks at around 779 nm, but with very small blue shift from MAPI-BOAs to MAPI-MAAc sample. Such shift may be related to the crystal size or the Cl doping in the MAPI-MACl in the films. It was also found that the MAPI-MACl sample has the largest band gap as shown in Fig. S5B.\(^\text{1}\) Besides, the peak width at half height of MAPI-BOAs film is slightly reduced which indicates its lower band-edge trap state compared to MAPI-MAAc film.\(^\text{27}\) Furthermore, the \( I-V \) response of the crystals in space-charge-limited-current (SCLC) regime is measured by the electron-only devices following the structure of FTO/TiO\(_2\)/perovskite layer/PCBM/Ag, as can be seen in Fig. 3C and D. At first stage of low voltage, the \( I-V \) response is ohmic. When the voltage increases, the \( I-V \) curve rises rapidly due to trap-filled limit (TFL), in which the injected charge carriers occupy all the trap states.\(^\text{28}\) The defect density is determined by the equation of \( N_t = \frac{2\varepsilon_0 V_{TFL}}{\varepsilon L^2} \), where \( \varepsilon \) is the dielectric constant of MAPbI\(_3\) (32), \( \varepsilon_0 \) represents vacuum permittivity, \( L \) is the thickness of perovskite layer (~260 nm, Fig. S2\(^\text{1}\)), \( e \) is the elementary charge. The density of trap states is calculated to be \( 5.2 \times 10^{15} \) and \( 1.3 \times 10^{16} \) cm\(^{-3}\) respectively for MAPI-BOAs and MAPI-MAAc samples. The calculated trap state density of MAPb-MACl is of \( 8.9 \times 10^{15} \) cm\(^{-3}\) (Fig. S6\(^\text{1}\)), which is even a little higher than that of the MAPI-BOAs film possibly due to the presence of the hole structures. The relative large \( N_t \) of MAPI-MAAc sample should be attributed to the smaller size of MAPI-MAAc sample’s crystals.

To further evaluate the performance of MAPI-BOAs perovskite, we then have measured its photoelectric properties. Fig. 4A shows the current (\( I \))–voltage (\( V \)) curves of the best performed device, which is measured under simulated AM 1.5 sunlight condition. The device’s PCE is 17.0% at reverse scan, and 16.2% at forward scan with negligible hysteresis, which indicates the balanced charge transport in this PSC.\(^\text{21,29}\) This should be associated with both the high quality and the growth orientation of perovskite crystals. For comparison, the MAPI-MAAc and MAPI-MACl PSCs were also measured and their devices performances were presented in Fig. S4.\(^\text{1}\) The MAPI-BOAs deliver the highest efficiency. The reason why MAPI-MACl device, which has very similar crystal quality with MAPI-BOAs, show relative low PCE, is that the morphology of MAPI-MACl contains too much pin-holes, which aggravates the electron–hole recombination. The improvement of the FF value of the MAPI-BOAs PSC than the MAPI-MAAc based device is attributed to the reduced surface and the grain boundary defects as verified in Fig. 3.
Fig. 4B is the external quantum efficiency (EQE) of the best devices and 88.3% incident photon-to-current efficiency is obtained. The integrated photocurrent density from the calculated photocurrent density of EQE spectrum is 20.6 mA cm$^{-2}$, which is very close to the short circuit current density measured by us.

Fig. 4C shows the superb stabilizing current/power output monitored by keeping the voltage at the maximum power point of 0.86 V. The reproducibility of MAPI-BOAs devices is also counted in Fig. S4.† In Fig. 4D, we measure the illumination stability of the PSCs under simulated AM 1.5 sunlight irradiation. The stability comparison is tested under one sun illumination with a relative humidity of 40%. During the interval of the I–V measurements, the devices were kept at open circuit condition to speed up the degradation of the device. We can see that the MAPI-BOAs device shows superior stability compared to the others under illumination situation. All these excellent photoelectric properties confirm the high quality of MAPI-BOAs perovskite crystals and the optimized film morphology of devices.

Conclusions

In summary, we have reported an easy and promising strategy of using blended organic additives to regulate the crystal growing process of perovskite for obtaining a balance between film coverage and crystallinity. The nucleation and crystal growth processes are masterly observed in this work by in situ microscopy measurement, which gives inspiration for better understanding of the mechanism of additives. By simply incorporating 20% MACl in MAAc additive, the morphology of the fabricated MAPbI$_3$ film is modified at no expense of crystal crystallinity. The density of trap states is highly reduced due to the increase of the grain size. The PCE of the PSCs fabricated by this strategy is obviously enhanced in comparison the ones using single additive. This work also provides us with a promising strategy to use different blended additives with different properties for further application in other perovskite systems.

Conflicts of interest

There are no conflicts to declare.

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