Coat-and-print patterning of silver nanowires for flexible and transparent electronics

Weiwei Li, Azat Meredov and Atif Shamim

Silver nanowires (Ag NWs) possess excellent optoelectronic properties, which have led to many technology-focused applications of transparent and flexible electronics. Many of these applications require patterning of Ag NWs into desired shapes, for which mask-based and printing-based techniques have been developed and widely used. However, there are still several limitations associated with these techniques. These limitations, such as complicated patterning procedures, limited patterning area, and compromised optical transparency, hamper the efficient fabrication of high-performance Ag NW patterns. Here, we propose a coat-and-print approach for effectively patterning Ag NWs. We printed a polymer-based ink on the spin-coated Ag NW films. The ink acts as a protective layer to help remove excess Ag NWs from the substrate and then dissolves itself into an organic solvent. In this way, we can take advantage of both coating-based techniques (lead to Ag NWs with high transparency) and printing-based techniques (efficiently pattern diverse shapes). The resultant Ag NW patterns exhibit comparable conductivity (sheet resistance: 7.1 to 30 Ohm/sq) and transparency (transmittance: 84 to 95% at λ = 550 nm) to those made by conventional coating methods. In addition, the patterned Ag NWs exhibit robust mechanical stability and reliability, surviving extensive bending and peeling tests. Due to higher conductivity, efficient patterning ability and inherent transparency, this material system and application method is highly suitable for transparent and flexible electronics. As a proof of concept, this research demonstrates a wide-band antenna, operating in the mm-wave range that includes the 5G communication band. The proposed antenna exhibits a wide bandwidth of 26 GHz (from 17.9 GHz to 44 GHz), robust return loss under 1000 cyclic bending (bending radius of 3.5 mm), and decent transparency over the entire visible wavelength (86.8% transmittance at λ = 550 nm). This work’s promising results indicate that this method can be adapted for roll-to-roll manufacturing to efficiently produce patterned and optically transparent devices.

npj Flexible Electronics (2019) 3:19; https://doi.org/10.1038/s41528-019-0063-3

INTRODUCTION

Transparent, flexible electronics have gained emerging attention in a variety of fields, including optoelectronics, energy storages, sensors, and light emitting displays.1-5 However, conventional conductors, such as indium tin oxide (ITO), have long suffered from high processing cost and lack of flexibility. Motivated by this issue, researchers have developed many ITO alternatives by using conductive materials with decreasing dimensions, including nanoscale metal mesh,6-10 conductive polymer,11,12 metal nanowires,13-15 graphene,16-21 and carbon nanotubes.22-24 Among these alternatives, silver nanowires (Ag NWs) are particularly attractive due to their excellent electrical conductivity and mechanical flexibility.25-29 In the past decade, tremendous efforts have been made to synthesize Ag NWs with high aspect ratio which is critical to the performance enhancement of Ag NWs-based devices. Thus far, various synthesis methods have been proposed, such as low-temperature,28 gentle reaction,29 multistep growth,30,31 and hydrothermal reduction,32 leading to ultrahigh aspect ratio of Ag NWs. Moreover, the resultant flexible Ag NW films have simultaneously demonstrated remarkable transparency and low sheet resistance,33-36 both of which are comparable and even superior to the ITO films.

Many of the Ag NW-based devices, such as electrodes, circuits, sensors, and thin film transistors, require Ag NW patterns on desired substrates.37-39 For this purpose, two types of techniques (i.e., mask-based and printing-based) are proposed and widely used for the patterning of Ag NWs. However, there are still several limitations to both methods. The mask-based patterning techniques, such as photolithography,39 stencil screen,39 wet/dry transfer,37 and ultraviolet/ozone (UV/O3) irradiation,41-43 are capable of producing highly transparent patterns, as this type of techniques can take advantage of state-of-the-art Ag NWs with high aspect ratio. However, mask use typically requires complex and high-cost processes that yield limited patterning area, as well as low patterning efficiency. By contrast, direct printing technologies offers roll-to-roll compatibility44-47 by which large-area, arbitrary conductive patterns have been achieved in cost-effective and time-effective way.38-50 However, the printing-based techniques only work for high-concentration inks or short Ag NWs (i.e., NWs with aspect ratios of less than 50).44,46,51 Therefore, the efficient manufacturing of high-performance Ag NW patterns, on which the success of Ag NW-based devices heavily hinges, remains challenging.

In this work, we introduce a coat-and-print method for manufacturing Ag NW patterns that offers a solution to the challenges mentioned above. Instead of directly printing Ag NWs, we pattern a polymer-based ink on the spin-coated Ag NW films with the aid of inkjet printing. This polymeric pattern acts as a protective layer that can help remove the excess Ag NWs from the substrate and then dissolves in an organic solvent. We show that the resultant Ag NW patterns can be as conductive and transparent as the coating-based or mask-based patterns and be...
fabricated as efficiently as the inkjet-printed ones. We further pattern the Ag NWs on a flexible substrate capable of sustaining thousands of bending and peeling cycles. The combination of our sample’s strong electrical performances and mechanical durability using the coat-and-print technique makes them particularly promising for the next-generation of flexible and transparent electronics. As a proof of concept, this research demonstrated a flexible and transparent antenna operating in the mm-wave range, including the 5G communication band.

RESULTS

The Ag NWs used for the patterning process had a average diameter of $27 \pm 9\, \text{nm}$, and length of $31 \pm 7\, \mu\text{m}$, which was confirmed by the scanning electron microscope (SEM) images presented in Fig. 1a and Fig. S1 in the Supporting Information. The typical X-ray diffraction (XRD) pattern of the Ag NWs exhibited four diffraction peaks indexed to the face-centered cubic Ag without any impurities. In addition, the intensity of peak (111) is much higher than those of (200), (220), and (311) due to the high aspect ratio of the Ag NWs, which is ideal for our coat-and-print method to build highly transparent and conductive Ag NW patterns. However, such long Ag NWs are hard to print with an inkjet printing technique, as the nozzle diameter (~$21.5\, \mu\text{m}$) is smaller than the length of Ag NWs, which may cause nozzle clogging.

A schematic of the freeform coat-and-print patterning process is presented in Fig. 1c. First, we deposited the Ag NW dispersion onto clean substrates, such as glass and plastic polymers to form a uniform and dense Ag NW layer; (ii) inkjet printing of polymer ink to form desired patterns; (iii) removing excess Ag NWs with wet tissues or tapes; and (iv) removing polymer by dissolving in organic solvents. Insets are digital photographs of the Ag NW ink and the PMMA ink used in the patterning process. LED light connected to the patterned transparent Ag NW circuits. The right panel displays the SEM images of the patterned Ag NW lines.

The transmittance ($T$) at $\lambda = 550\, \text{nm}$ as a function of sheet resistance ($R$) of the Ag NW films fabricated by coating, and coat-and-print technique, as well as previously developed methods to prepare transparent electrodes. Inset depicts the photograph of one representative Ag NW patterns on glass ($T = 95\%$, $R = 30\, \text{Ohm/sq}$) in front of KAUST logo.
PMMA layer by immersing the sample into acetone for 5 min and completed the patterning processes (Fig. S2c). What should be noted is that the etching time to dissolve the PMMA layer highly depends on the solubility of PMMA in organic solvents and the bath temperature. We can decrease the etching time to less than 30 s in anisole and N,N-dimethylformamide (DMF) bath at 60 °C (Fig. S2d, e). Figure 1d demonstrates typical Ag NW patterns on a flexible polyethylene terephthalate (PET) substrate, where fine, micro-scale structures are designed and patterned on the Ag NW networks as displayed in the magnified SEM images. These patterning steps can be easily repeated on other substrates, such as four-inch fused silica glass, 50 × 50 mm² flexible PET and 50 × 50 mm² Kapton film, as depicted in Fig. S3 in the Supporting Information, which demonstrates the patterned KAUST logo, letters, serpentine lines, straight lines, capacitors, circles, stars, and other complex drawings with clear edges and boundaries.

We further studied the electrical and optical performance of various coated and patterned Ag NW films, which were controlled by changing the coating cycles (one to five cycles) and Ag NW concentration (0.5–1.5 mg/mL). The transmittance of the Ag NW films changed from 84.2 to 95.6% when sheet resistance increased from 7.1 to 36 Ohm/sq (square markers in Fig. 1e). As illustrated in Fig. 1e, our coat-and-print samples (red markers) are nearly identical to the originally coated ones (blue markers) in both conductivity and transmittance, regardless of the Ag NW films used. This result implies that the printed PMMA has been removed during the patterning process and the extra processing in our coat-and-print method causes no noticeable performance degradation to the Ag NW films, indicating the stability and reliability of the proposed patterning technique. Moreover, to achieve the same sheet resistance, our coat-and-print Ag NW patterns feature much higher optical transmittance than the patterns by previously reported printing-based methods (Fig. 1e). The reason is attributed to the ultrahigh aspect ratio of the Ag NWs (~1000) used in this work, whereas only short nanowires (aspect ratio < 50) can be used in the reported inkjet printing process. Moreover, Ag NW films patterned by our method demonstrates comparable electrical and optical performance to the reported methods, such as photo mask patterning, pattern mask patterning, inkjet printing, micro-contact printing, shadow mask patterning, and magnetic mask printing, as well as other transparent electrodes, such as seed particle printing of metal alloys and laser writing of Ag nanoparticles grid. We thus suggest that our coat-and-print method can take advantage of both coating-based techniques (can lead to patterns with high conductivity and transparency) and printing-based techniques (can efficiently pattern diverse shapes).

Fig. 2  a SEM images of the patterned dots (diameter 60 μm), lines (80 μm in spacing), and cross lines (60 μm in width). All the unmarked scale bars are 100 μm. b The line thickness of the patterned Ag NWs as a function of the printing passes. c The line width plotted as a function of the interdrop spacing. Insets are SEM images of the patterned Ag NW lines. Scale bars are 100 μm. d Measured resistance with different line length and various width from three separate Ag NW patterns. e Calculated sheet resistance plotted as a function of line width for the patterned Ag NW lines. f Thickness and calculated electrical conductivity as a function of line width for the patterned Ag NW lines.
To further highlight the cost-effective and time-effective advantages of our coat-and-print method, we demonstrated the patterning of Ag NWs in diverse shapes. Figures 2a and S4 display the patterned Ag NW-based dots (diameter in 60 μm), lines (width in 60 μm and spacing in 80 μm), cross lines (width in 60 μm), and squares, circles, and complex drawings with clear edges and boundaries. Essentially, the coat-and-print method is accessible to arbitrary patterns by designing the digital files in commercial software, such as AutoCAD and CorelDraw, or the pattern editor built into the printer software. Notably, the thickness and...
transparency of these patterns are controlled by the concentration of the Ag NW dispersion and coating parameters, and the spatial resolution can be guaranteed by the inkjet printing.

Then, we focused on the basic line patterns and explored how the line parameters affect the patterns’ overall electrical conductivity. One important parameter in the coat-and-print method is the printing passes of the PMMA layer: the width of the patterned lines only fluctuated between 86 µm and 90 µm as the number of printing passes increased (Fig. S5) while the resultant thickness was significantly influenced (Fig. 2b). Specifically, the thickness of the Ag NW lines increased from ~130 nm to ~420 nm and became stable (~420 nm) after three printing passes. This is because that PMMA layers obtained by one or two printing passes were too thin, and thus could not provide sufficient protection for Ag NWs. As a result, some Ag NWs were removed during the follow up wiping process. In contrast, three printing passes were enough to print protective layers to pattern the Ag NWs. In Fig. 2c, we illustrated that the line width were easily controlled by the drop spacing. A drop spacing in the inkjet printer means the deposition distance of two droplets. When the drop spacing is larger, the distance between two deposited ink droplets increases, which results in a line with narrower width. For example, the patterned line width decreased from 180 µm to 49 µm when the drop spacing increased from 10 µm to 50 µm. SEM images (inset of Fig. 2c) reveal the line width variation. It can also be seen that the widths of these lines are quite uniform. We then investigated the detailed electrical performance of the patterned lines with line widths ranging from ~0.1 mm to ~4 mm, and lengths ranging from 5 mm to 30 mm. Figure 2d plots the measured resistance of the Ag NW lines where error bars are created by three time measurements on three separate samples with the same coating and printing parameters. We found that the resistance increases linearly with the line length for all samples, implying the uniformity of the patterned lines using our proposed patterning technique. Quantitatively, sheet resistance (Rs) of the printed lines was calculated according to the measured resistance (Rl), line width (w) and line length (l) through $R_s = R_l / w$. The calculated Rs for samples with various line widths is displayed in Fig. 2e. The sheet resistance was relatively stable (fluctuating between 2.8 Ohm/sq to 3.3 Ohm/sq) regardless of the line width. These results again prove the uniformity of the patterned Ag NWs, which is particularly essential for the reliability of Ag NW-based electrical devices. We then characterized the electrical conductivity (σ) of the patterned Ag NW lines using line width as a variable by calculating $\sigma = 1 / (t \cdot Rs)$, where $t$ is the thickness of Ag NW lines. As Fig. 2f illustrates, for the patterned lines with different line widths, the electrical conductivity had a small range (~7–8 KS/cm), again demonstrating the excellent repeatability of the proposed patterning technique.

Having characterized the excellent electrical conductivity and uniformity of the patterned Ag NW lines, we then evaluated the electrical/mechanical stability. The patterned Ag NW lines were firstly subjected to cyclic bending with different radii of curvature and measured their resistance change. Figure 3a demonstrates that the resistance of the patterned Ag NW lines with various widths can nearly remain constant under a bending radius as low as 3.5 mm. Such durability is further highlighted in Fig. 3b where only slight changes in the resistance are observed for patterned lines after 1000 bending cycles. The pristine Ag NW lines are easy to fail in conductivity after performing tape test (Fig. 3c). To further improve their resistance to wear and friction, 0.5% polyvinyl alcohol (PVA) solution (water: ethanol = 1:1, weight ratio) was spin-coated on the surface of the patterned Ag NW lines. Consequently, the resistance of the patterned line was largely maintained (the relative increase is ~20%) after 1000 tape-peeling tests, as depicted in Fig. 3c.

The mechanical stability of our patterned Ag NWs, combined with their excellent conductivity and transparency, makes them particularly attractive for flexible electronic devices. To demonstrate the potential applications of the proposed coat-and-print technique for patterning Ag NWs, a flexible, transparent monopole antenna for the applications of mm-wave, including 5G communication was fabricated and characterized. Figure S6 displays the geometric parameters of the proposed radio frequency (RF) antenna and the 3D model in the electromagnetic simulation software (CST Microwave Studio). We used a triangular monopole antenna with slots (width: 150 µm) and a slotted ground layer (to improve transparency), sandwiched between a low-loss dielectric substrate (tanδ = 0.0044 at 24 GHz) with a thickness of 50 µm. The total size of the antenna is 12 x 12 mm². Figure 4a presents the simulated (dashed line) and measured (solid line) reflection coefficient (S11) of the monopole antenna, where we found a match in the range of 17.9 to 40 GHz. The ~10 dB bandwidth covers the whole measurement range, indicating an ultra-wide bandwidth, which is beneficial for the applications of 5G wireless communication. To further evaluate the radiation performance of the fabricated Ag NW monopole antenna, the realized antenna gain was simulated and is depicted in Fig. 4b. The Ag NW antenna had a reasonable gain, varying from ~1.7 to 1.2 dBi from 18 to 40 GHz, though the gain was lower than the same antenna made by copper with a conductivity of 5.8 x 10⁷ S/m and a maximum gain of 3.5 dBi (Fig. S7 in Supporting Information).

We conclude our work by highlighting the robustly mechanical flexibility and excellent optical transparency of the fabricated antenna. As displayed in Fig. 4d, the measured S11 is almost constant under the bending radii of 20 mm and 5 mm. To further prove the robustness of the monopole antenna under deformation, repetitive bending cycles with the radius of 3.5 mm were performed, and the reflection coefficient of the antenna was recorded after 1000 cycles. Figure 4c confirms that the ~10 dB bandwidth performed robustly over a range of 22 GHz without any significant degradations, despite the S11 slightly shifting. This result implies that the antenna has an excellent ability to maintain conductive pathways under cyclic deformations. Furthermore, the transmittance of the monopole antenna is presented in Fig. 4d, with a flexible Ag NW antenna proves its strong optical transparency with more than 70% transmittance over the entire visible wavelength range (400–800 nm). Transmittance at the wavelength of 550 nm is 86.8%, which is higher than the reported transparent antennas, such as 75% for micro-metal mesh conductive film-based antenna,56 80% for Ag-coated polyester-based antenna,57 and 85% for graphene-based antenna.58

**DISCUSSION**

In this paper, we presented a reliable, low-cost coat-and-print technique to pattern Ag NWs with the assistance of inkjet printing. Attributed to the thin and long Ag NWs synthesized in this study, the Ag NW films exhibited excellent conductivity and transmittance compared to the printed transparent electrodes in the literature. Various shapes with different sizes were successfully patterned on rigid glass and flexible substrates with large area and high precision. As a potential application of the proposed patterning approach, flexible and transparent RF antenna was demonstrated and experimentally verified. The antenna had an ultra-wide bandwidth (22 GHz of the ~10 dB bandwidth), reasonable antenna gain (maximum gain: 1.2 dB), excellent robustness (more than 1000 bending cycles with a radius of 3.5 mm), and high optical transparency (more than 80% transmittance over the visible range). We believe that the proposed coat-and-print technique can enable fabrication of reliable, precise Ag NW patterns in time-effective and cost-effective ways, and can be extended to other nanomaterials, such as nanoparticles and copper nanowires.
METHODS

Synthesis and purification of Ag NWs

The Ag NWs with ultrahigh aspect ratios were synthesized using a modified polyol reduction process. First, Ag nitrate (0.12 M, 10 mL), polyvinyl pyrrolidone (PVP, 60 mM, 60 mL), and cetyltrimonium bromide (CTAB, 1 mL) were prepared in ethylene glycol at room temperature as three separate solutions. In a typical synthesis procedure, the CTAB solution was added to the PVP solution with vigorous stirring for 1 min. Then, the mixed solution was transferred to a reactor preheated to 160 °C. After gently stirring for 30 min, the Ag nitrate solution was injected dropwise into the mixed solution at a jetting rate of 1 mL/min. Another 60 min were needed to complete the reaction. Afterward, two-times volume of acetone was added to the cooled dispersion to precipitate the Ag NWs. The product was re-dispersed into water and washed with acetone three times until the supernatant was clear. Then, the product was washed another two times with ethanol to replace the solvent, and re-dispersed into ethanol with concentrations of 1.5 mg/mL, 0.75 mg/mL, and 0.5 mg/mL. Finally, the dispersion was centrifuged at low speed (~600 rpm for 1 min) to obtain well-dispersed Ag NW dispersion by pipetting the supernatant.

Preparation of conductive Ag NW films

Four-inch glass with a thickness of 1 mm, 125-μm-thick PET and 50-μm-thick Kapton were used as substrates in the experiment. First, the substrates were carefully cleaned with deionized water and ethanol in the sonication bath. Then they were dried with nitrogen. The obtained Ag NW dispersion was deposited on the cleaned substrates by drop coating or sonication bath. Then they were dried with nitrogen. The obtained Ag NW films were molded as a discrete 50-Ohm port. The PMMA powder with a molar ratio of 120,000 was dissolved in a mixed polyol solution with a 1:3 ratio of water and polyvinyl alcohol (PVA, 30 wt%). Afterward, 1.5 mL of the freshly prepared PMMA was filtered through a 0.2 μm nylon filter and then injected into the cartridge. A commercial piezoelectric inkjet nozzle (Dimatix) with a diameter of 21 μm and a drop volume of 10 pl was used to inject the PMMA solution onto prepared Ag NW films. The temperature of platen was set at 45 °C to guarantee the uniform spreading of PMMA drops on Ag NW films. Different patterns were designed either through the Dimatix pattern editor or the commercial software Coreldraw and then directly printed. After depositing the PMMA protective layer, the sample was placed into an oven at 110 °C for 5 min to evaporate the solvents. Next, wet tissue was used to wipe the excess Ag NWs. Finally, the sample was immersed in acetone for 5 min to dissolve PMMA, followed by drying naturally to complete the patterning of Ag NWs.

Simulation and fabrication of the RF antenna

Before the fabrication, the antenna structures were optimized and the optimal parameters were simulated in CST Microwave Studio to verify the performance. The monopole antenna was simulated through a thin surface impedance layer with a thickness of 500 nm and a conductivity of 7 × 10^5 S/m. A 50-μm-thick layer with a dielectric constant of 2.6 and loss tangent of 0.0044 at 24 GHz was used as the dielectric substrate. The source was modeled as a discrete 50-Ohm port.

To fabricate the monopole antenna, Ag NW dispersion was coated on a clean Preflex substrate (thickness: 0.05 mm), dried naturally, and patterned using the method described above. Then, the ground layer was patterned with the same method. Finally, the fabricated antenna was baked in an oven at 100 °C for 10 min.

Characterization

The SEM images of the synthesized Ag NWs and the patterned Ag NW films were taken using a scanning electron microscope (FEI NovaNano FEG-SEM 630) after sputtering 3 nm Ir (Quorum Q150TS) to increase conductivity. The crystallinity of the Ag NWs was examined using XRD (Bruker D2 PHASER) in the range of 30–80° with a scanning intercept of 0.12 degree/s. The UV-vis absorption spectrum of the Ag NW dispersion and prepared Ag NW films were obtained using a UV-vis spectrophotometer (Thermo Evolution 600). The sheet resistances of the Ag NW films were measured using a four-point probe system (CMT-SR2000N). A vector network analyzer (Agilent N5225A) was used to obtain S_11 of the fabricated antenna.

DATA AVAILABILITY

The datasets generated during and/or analyzed during this study are available from the corresponding author on reasonable request.

Received: 9 August 2019; Accepted: 8 November 2019; Published online: 10 December 2019

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