Antifouling Coatings Generated from Unsymmetrical Partially Fluorinated Spiroalkanedithiols

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ABSTRACT: Biofouling negatively impacts modern society on a daily basis, especially with regard to the important industries of medicine, oil, and shipping. This manuscript describes the preparation and study of model antifouling coatings generated from the adsorption of unsymmetrical partially fluorinated spiroalkanedithiols on gold. The antifouling properties of the self-assembled monolayers (SAMs) derived from the spiroalkanedithiols were compared to SAMs derived from analogous monodentate partially fluorinated and nonfluorinated alkanethiols. The antifouling properties were evaluated using in situ surface plasmon resonance spectroscopy (SPR), ex situ electrochemical quartz crystal microbalance (QCM) measurements, and ex situ ellipsometric thickness measurements. The resistance to nonspecific protein adsorption of the SAMs was evaluated with proteins having a wide range of properties and applications including protamine, lysozyme, bovine serum albumin, and fibrinogen. The results from the SPR and the QCM measurements demonstrated that in most cases, the SAM coatings derived from the partially fluorinated spiroalkanedithiols having mixed hydrocarbon and fluorocarbon tail groups exhibited better antifouling performance when compared to the SAMs derived from their single-component monodentate counterparts. The studies also revealed that while the SPR and the QCM measurements in most cases were able to distinguish the adsorption trends for the SAMs and proteins examined, the ellipsometric thickness measurements were markedly less discriminating. On the whole, these studies validate the use of unsymmetrical partially fluorinated spiroalkanedithiols for generating effective antifouling coatings on metal substrates.

KEYWORDS: antifouling, self-assembled monolayers, fluorinated, spiroalkanedithiols, surface plasmon resonance, quartz crystal microbalance, ellipsometry

INTRODUCTION

Biofouling or biocontamination creates challenges in various applications, including biosensors, medical implants, surgical instruments, marine equipment, cookware, protective apparel, and food packaging. In medical devices, bioresistance can interfere with device performance, product efficiency, and customer safety. Moreover, uncontrolled adhesion of biomaterials (e.g., proteins) on implanted medical devices can diminish the effectiveness of the device and affect the patient’s health. Consequently, research on novel antifouling coatings remains an active area of research in materials and interfacial science. To this end, organic thin films bearing polyethylene glycol (PEG) termini have been widely used as coatings for biomedical applications. While exhibiting a high degree of biological inertness, PEG-based coatings suffer from the reactivity of the PEG molecules toward hydration at high temperatures, as well as oxidation by atmospheric oxygen in the presence of transitional metal ions. Ubiquitous polytetrafluoroethylene (PTFE), commonly known as Teflon, is an alternative antiadhesive material whose repeat units have been incorporated into thin-film coatings. PTFE-based films are well-recognized antiahesive materials with desirable interfacial properties, such as low wettability, low friction, and low adhesion. Consequently, the introduction of fluorocarbon segments in nanostructured thin films (1–2 nm thickness) can lead to materials with unique interfacial properties that limit the biofouling of surfaces. Thin-film coatings generated using the process of self-assembly afford well-defined, highly ordered surfaces known as self-assembled monolayers (SAMs). A wide variety of SAMs with distinct interfacial properties have been generated from systematically designed adsorbates for use in targeted applications as well as fundamental studies of interfacial science. SAMs generated by the adsorption of thiols on
gold surfaces serve as a particularly versatile system for studying the interfacial properties of films, offering unprecedented insights into the development of effective antifouling surfaces.\textsuperscript{15,28–34} SAMs on gold enjoy several advantages including high reproducibility, wide functional group tolerance, and interfacial tunability via the use of adsorbates bearing selected tail groups.\textsuperscript{12,15,16,28} Well-established chemical reactions have been used to synthesize adsorbates bearing PEG, charged, or zwitterionic tail groups for generating a variety of antifouling surfaces.\textsuperscript{25,29–34} Furthermore, these tools have allowed for the synthesis of partially fluorinated adsorbates to generate thin films with interfacial properties similar to those of PTFE. Thus, similar to PTFE, fluorinated SAMs (FSAMs) have been shown to exhibit high hydrophobic and oleophobic behavior, as well as chemical and thermal stability.\textsuperscript{2,16,35–37} The incorporation of fluorinated termini in the structure of thiol-based adsorbates bridges the gaps in applications where the use of fluorinated polymers is inappropriate. Research on fluorinated SAMs has shown that the structure of the film is greatly affected by the hydrocarbon spacer of the adsorbates, whereas the fluorinated segments dictate the interfacial properties and thermal stability of the film.\textsuperscript{16–18,35–37}

An emerging method for tuning the interfacial properties of thin-film coatings is the use of mixed adsorbates bearing dissimilar functional groups to generate unique nanoscale interfaces composed of phase-incompatible chemical entities.\textsuperscript{30,38–40,45} However, the incorporation of two different monodentate thiols possessing chemically dissimilar tail groups often leads to films comprised of phase-separated domains due to their incompatibility.\textsuperscript{41–44} Only recently, Chinwangso and co-workers demonstrated the ability to generate films with controlled interfacial heterogeneity by linking two chemically dissimilar chains—a hydrocarbon chain with a partially fluorinated chain or an oligo(ethylene glycol) chain—on a bidentate spiroalkanedithiol (SADT) head group, as illustrated in Figure 1.\textsuperscript{30,38–40,45}

Inspired by the unique antiadhesive properties as well as chemical and thermal stability of fluorinated SAMs and spiroalkanedithiol-based SAMs,\textsuperscript{26,27,30,36} this manuscript describes the interactions between several common contaminant proteins and compositionally mixed interfaces generated from partially fluorinated spiroalkanedithiols. The overarching goal of this work is to generate interfacially “conflicted” monolayers on gold surfaces that resist protein adhesion. We use the term...
“conflicted” to emphasize that the interfaces are comprised of chemically disparate species that are held in close proximity while preferring to be phase-separated. Specifically, we generated SAMs from unsymmetrical partially fluorinated spirosilanealkanethiols (SADTs), F8H10-C12 and F8H10-C18 (see Figure 2), to study protein adhesion on these unique compositionally heterogeneous surfaces. The performance of the partially fluorinated SADT-based SAMs was compared to SAMs generated from the analogous monodentate adsorbates n-alkanethiol C16SH and partially fluorinated alkanethiol F8H8SH to evaluate the effect of the adsorbate structure on the antiadhesive properties of the films. Given their wide range of sizes, structures, and chemical compositions, we chose protamine, lysozyme, bovine serum albumin (BSA), and fibrinogen to serve as model proteins to evaluate the antiadhesive properties of the “conflicted” interfaces.

We hypothesize that the chemical heterogeneity introduced at the interfaces of the bis-functionalized spirosilanealkanethiol SAMs can plausibly lead to a reduction in favorable interactions between the contacting proteins and the surfaces as the SAMs are composed of disparate low-energy species. We also expect that the perfluorinated segment will help maintain the structural integrity and increase the thermal stability of the films. Notably, adsorbate F8H8-C12 is designed to create SAMs that allow the bulky helical fluorinated segments to pack atop the underlying well-packed trans-extended alkyl chains. We characterized the protein-resistant properties of the single-component and mixed SAMs by surface plasmon resonance (SPR), electrochemical quartz crystal microbalance (QCM), and ellipsometry. We anticipate that studies of these model SAM interfaces will guide the development of “conflicted” films as nanoscale antiadhesive coatings that can be fine-tuned to enhance the biocompatibility of medical implants and devices but also in the ever-important oil and shipping industries.

### EXPERIMENTAL SECTION

The Supporting Information provides details regarding the materials, instrumentation, and procedures used to conduct the research in this manuscript (including details used to collect the SPR and QCM data). The adsorbates and SAMs utilized in the present study were fully characterized in previous reports.

### RESULTS AND DISCUSSION

We first established an initial, but not exhaustive, selection of testing environments to evaluate antifouling properties of the SAMs by examining proteins with a wide range of properties and applications. Proteins evaluated in this study were selected based on criteria including isoelectric point, size, molecular weight, hydrophobicity, and applications. Table 1 provides a description of the properties of the selected proteins. The set of proteins includes protamine, which was selected due to its applications in medicine and tissue engineering; lysozyme, a small and positively charged protein; BSA, a widely used, stable, and hydrophobic protein; and fibrinogen, a widely used large and sticky protein. All protein solutions were prepared by dissolving the protein in phosphate buffer solution (PBS) since it is well-tolerated by the selected proteins. Furthermore, to remove any anomalies in protein adhesion that might be caused by the buffer system, we measured changes associated with the buffer exposed on each respective film and used this information as a reference for the protein experiments. We studied the nonspecific adsorption of the selected proteins on the surfaces of SAMs generated from unsymmetrical partially fluorinated spirosilanealkanethiols (SADTs), F8H10-C12 and F8H10-C18, along with monothiolate SAMs formed from a normal monodentate n-alkanethiol C16SH and a partially fluorinated alkanethiol F8H8SH. Such comparisons allow detailed structure–property relationships of the interfacially “conflicted” monolayers toward antifouling. Moreover, we also measured qualitatively both the surface coverage and the amount of protein on the SAM surfaces using surface plasmon resonance (SPR), electrochemical quartz crystal microbalance (QCM), and ellipsometry.

Together with ellipsometry, a common and easily accessible method, we chose to explore the use of other, more sensitive biophysical techniques, such as SPR and QCM, to examine the protein resistance of thin films derived from the adsorbates, as shown in Figure 2. The biophysical methods are needed not only to evaluate the validity of the ellipsometry data but also to gain broader insight into the antifouling performance of the unique “conflicted” interfaces prepared herein. Importantly, both QCM and SPR provide data that are highly correlated to mass absorbance on the surfaces; consequently, we report in Figure 3 the adsorption behavior of four different proteins on the SAMs as determined by data obtained from SPR, QCM, and ellipsometry.

Before describing the results of each type of measurement (SPR, QCM, and ellipsometry) in detail, we note that the antifouling trends were most consistently demonstrated by the adsorption of the largest-molecular-weight protein fibrinogen (340 KDa) across all of the SAMs, where an overall picture emerged that the SAMs derived from the unsymmetrical partially fluorinated spirosilanealkanethiol adsorbates were able to reduce the nonspecific absorption of fibrinogen. Specifically, the SPR values for fibrinogen on the C16SH and F8H8SH SAMs were 2531 ΔRU and 2533 ΔRU, respectively, which trended downward to 2327 ΔRU and 2344 ΔRU on the F8H10-C18 and F8H10-C12 SAMs, respectively (Figure 3A). Likewise, the QCM data showed large values of increased mass upon fibrinogen exposure, 792 and 793 ng/cm² for the C16SH and F8H8SH SAMs, respectively, which trended downward to 565 and 571 ng/cm² for the F8H10-C18 and F8H10-C12 SAMs, respectively (Figure 3B). These observations were
corroborated by the ellipsometry data (Figure 3C), where the greatest increases in thickness were 37 and 45 Å for the C16SH and F8H8SH SAMs, respectively, compared to 27 and 30 Å for the F8H10-C12 and F8H10-C18 SAMs, respectively.

**Figure 3.** Inferred changes in mass/thickness for SAMs generated from C16SH, F8H8SH, F8H10-C12, and F8H10-C18 upon exposure to solutions of protamine, lysozyme, BSA, and fibrinogen as measured by (A) SPR, (B) QCM, and (C) ellipsometry. The error bars were generated from measurements on three independent samples of each surface and protein.

**In Situ Analysis of Protein Adhesion Using SPR Spectroscopy.** Surface plasmon resonance (SPR) spectroscopy is a convenient technique to monitor the real-time *in situ* interaction of proteins with surfaces in both academic and
SPR is an optical technique that detects changes in the refractive index as the material adheres to a surface. Here, changes in response units are correlated to the amount of nonspecific proteins that adsorb on SAMs. Increasing material deposition generates increasingly large changes in response units (ΔRU). Figure 4 displays sensorograms reporting the response changes for the interactions between four selected proteins and the SAMs, while Table 2 shows the numerically calculated ΔRU values.

The SPR data in Figure 4 and Table 2 show generally that the SAMs derived from C16SH and F8H8SH absorbed larger amounts of proteins compared to the SAMs derived from F8H10-C12 and F8H10-C18, particularly for the F8H10-C12 SAMs. For example, protamine exposure produced ΔRU values of 544, 461, 233, and 453 for SAMs generated from C16SH, F8H8SH, F8H10-C12, and F8H10-C18, respectively. The value of ΔRU correlates to the amount of material interacting and binding to the surface of the SAM attached to the gold substrate. More protein binding on the surface would generate a greater response signal. Thus, a decrease in response units is expected with respect to the decrease in the amount of proteins on the surface. The data for protamine correspond to 15, 48, and 17% less protein adsorption on the surface of SAMs generated from F8H8SH, F8H10-C12, and F8H10-C18, respectively, compared to the C16SH SAMs (normalized at 100% protein adsorption). In this case, the F8H10-C12 SAMs exhibited greater protein resistance than those generated from analogous F8H10-C18. Interestingly, when comparing the ΔRU values in Table 2 for the bidentate F8H10-C12 and F8H10-C18 SAMs upon exposure to lysozyme, BSA, and fibrinogen, these two SAMs showed no substantial differences in antifouling behavior; however, their ΔRU values were notably smaller than those found for the monodentate C16SH and F8H8SH SAMs.

Overall, the SAMs generated from the bidentate adsorbates (F8H10-C12 and F8H10-C18) exhibited greater protein resistance than the SAMs generated from the monodentate analogues (C16SH and F8H8SH). Due, at least in part, to its hydrophobic nature (see Table 3), the C16SH SAM can interact strongly with hydrophobic patches on the proteins to afford large ΔRU values. Compared to alkanethiol SAMs, fluorinated surfaces exhibit even greater hydrophobicity and lower surface energies. The hydrophobicity and low surface energy of the fluorinated surface give rise to unfavorable interactions with the proteins (i.e., repulsive interactions) when compared to the C16SH SAM. Specifically, for the F8H8SH SAM, the perfluorinated portion produces a hydrophobic surface with lower surface energy than alkanethiol SAMs. In the case of the F8H10-C12 SAM, greater protein resistance is likely due to the ability of the hydrophobic helical F8 fluorinated chains (van der Waals diameter of ∼5.6 Å) to extend above the underlying well-packed trans-extended alkyl chains (van der Waals diameter ∼4.2 Å). Specifically, inclusion of the C12 alkyl chains in the F8H10-C12 SAM architecture allows the F8 fluorinated chains to be more loosely packed than those in the monodentate corresponding SAM (F8H8SH), thereby exposing greater numbers of antiahesive interfacial CF2 groups per unit area than in the F8H8SH SAMs.

In contrast to the F8H10-C12 SAMs, the longer hydrocarbon chains in the F8H10-C18 SAMs were found in previous studies to be detrimental to film order; this phenomenon was attributed to unfavorable interactions between the two phase-incompatible groups that led to loosely packed chains exposing both CH2 and CF2 groups. Nevertheless, the SAMs derived from bidentate F8H10-C18 were, on the whole, more protein-resistant than the SAMs derived from the monodentate analogues. Consequently, we attribute the enhanced protein resistance of the F8H10-C18 SAMs to the heterogeneous mixture of interfacial hydrocarbon and

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**Figure 4.** SPR sensorgrams of the SAMs exposed to (A) protamine, (B) lysozyme, (C) BSA, and (D) fibrinogen. Protein solutions were prepared at a concentration of 1 mg/mL in a PBS. The measurements were performed on three independent samples of each surface and protein with all trends consistent with those shown here.
fluorocarbon species, which represent an unnatural composition that is not found in nature.

Importantly, there is no convincing evidence that the size/molecular weight of the proteins has an influence on the $\Delta$RU values; furthermore, the net charge of the proteins also seems to play no role on the amounts of protein adsorbed on the SAM surfaces. Similarly, there is no clear correlation with the surface energies of the SAMs: $\text{C}_{16}\text{SH}$ SAM (20.3 mJ/m$^2$), $\text{F}_{8}\text{H}_{8}\text{SH}$ SAM (8.9 mJ/m$^2$), $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$ SAM (11.5 mJ/m$^2$), and $\text{F}_{8}\text{H}_{10}-\text{C}_{18}$ SAM (15.8 mJ/m$^2$). Details of the surface energy calculations for the SAMs can be found in the Supporting Information (see Tables S2–S4). Given these observations, it is likely that, when comparing the respective proteins, the relative $\Delta$RU values on these uniformly low-energy SAM surfaces are predominantly influenced by the sticky nature of the respective proteins, with the stickiest protein of all (fibrinogen),$^{49,50}$ showing the largest $\Delta$RU values.

**Ex Situ Analysis of Protein Adhesion Using QCM.**

Taking advantage of the mass sensitivity of QCM sensors,$^{47,48,64,65}$ we also used QCM to quantify the amount of protein adhered to the SAM surfaces. Figure 5 shows the frequency change $\Delta f$ as a function of time for all four SAM surfaces after 1 h exposure to 1 mg/mL of proteins in PBS; a decrease in frequency compared to the bare QCM sensor indicates the mass adsorbed onto the surface. The corresponding mass loadings of each protein on the surfaces (calculated using the Sauerbrey equation) are listed in Table 4. Each reported value is an average of three independent experiments. Notably, the data show greater mass changes after exposing protein solutions to the $\text{C}_{16}\text{SH}$ and $\text{F}_{8}\text{H}_{8}\text{SH}$ SAMs compared to the $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$ and $\text{F}_{8}\text{H}_{10}-\text{C}_{18}$ SAMs, indicating a lesser mass loading on the two bidentate SAMs. The QCM studies were conducted ex situ and therefore generated responses known as dry mass loadings of proteins on the surface without the hydration layer, which should plausibly reflect lower protein loadings when compared to the data obtained from in situ measurements by SPR.$^{66-69}$ Nevertheless, similar trends would be expected from both methods.

As noted when analyzing the SPR data, the enhanced protein resistance observed by QCM for the $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$ SAMs can be attributed to the extension of the hydrophobic helical $\text{F}_8$ fluorinated chains above the underlying densely packed alkyl chains, thereby exposing greater numbers of antiadhesive interfacial CF$_2$ groups per unit area than in the $\text{F}_{8}\text{H}_{8}\text{SH}$ SAMs. Similarly, the enhanced protein resistance

| adsorbate     | protamine $\Delta$RU | lysozyme $\Delta$RU | BSA $\Delta$RU | fibrinogen $\Delta$RU |
|---------------|-----------------------|----------------------|----------------|-----------------------|
| $\text{C}_{16}\text{SH}$ | 544 ± 110             | 1407 ± 16            | 948 ± 28       | 2531 ± 122           |
| $\text{F}_{8}\text{H}_{8}\text{SH}$ | 461 ± 49              | 1302 ± 153           | 968 ± 116      | 2533 ± 119           |
| $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$ | 233 ± 30              | 1084 ± 18            | 833 ± 84       | 2344 ± 110           |
| $\text{F}_{8}\text{H}_{10}-\text{C}_{18}$ | 453 ± 19              | 1059 ± 14            | 847 ± 69       | 2327 ± 63            |

Table 3. Advancing Contact Angles ($\theta_a$, $^\circ$) for Water and Hexadecane as Probing Liquids on the SAM Surface

|            | water (H$_2$O) $\theta_a$ $^\circ$ | hexadecane (HD) $\theta_a$ $^\circ$ |
|------------|-------------------------------------|-------------------------------------|
| $\text{C}_{16}\text{SH}$ | 108                                | 51                                  |
| $\text{F}_{8}\text{H}_{8}\text{SH}$ | 120                                | 83                                  |
| $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$ | 125                                | 73                                  |
| $\text{F}_{8}\text{H}_{10}-\text{C}_{18}$ | 121                                | 59                                  |

Figure 5. Change in frequency vs time for SAMs derived from $\text{C}_{16}\text{SH}$, $\text{F}_{8}\text{H}_{8}\text{SH}$, $\text{F}_{8}\text{H}_{10}-\text{C}_{12}$, and $\text{F}_{8}\text{H}_{10}-\text{C}_{18}$ after 1 h exposure to 1 mg/mL of protein in PBS solution: (A) protamine, (B) lysosome, (C) BSA, and (D) fibrinogen.
observed for the F8H10-C18 SAMs can be attributed to unfavorable interactions between the two phase-incompatible hydrocarbon and fluorocarbon tail groups that give rise to a loosely packed mixture of chains that expose both CH$_2$ and CF$_2$ groups—mixtures that are not found in nature and can plausibly lead to diminished protein adsorption.

The QCM data in Figure 5 and Table 4 also offer no convincing evidence that the size/molecular weight of the proteins, the net charge of the proteins, or the surface energies of the SAMs influence protein adsorption in a systematic manner. These data provide further support for our hypothesis that when comparing the respective proteins, the relative degree of protein adsorption is predominantly influenced by the sticky nature of the respective proteins, exhibiting the greatest degree of adsorption on the low surface energy SAMs.

Ellipsometric Thickness Measurements. The ellipsometric thicknesses for all SAMs examined are provided in Table S1 in the Supporting Information, where the data confirm monolayer formation and are consistent with literature values. Figure 3C graphically presents the changes in thickness after protein exposure, and Table 5 lists the numerical thickness values. As noted above, the adsorption data for fibrinogen are consistent with the trends observed by SPR and QCM, namely, the SAMs generated from bidentate F8H10-C12 and F8H10-C18 were more resistant to the adsorption of fibrinogen than the SAMs generated from monodentate C16SH and F8H8SH. This trend was not evident in the adsorption behaviors of protamine, BSA, and lysozyme on the SAMs as evaluated by ellipsometry. Specifically, the ellipsometric thicknesses upon protein exposure were largely within the experimental error on all SAMs for each protein examined, save for fibrinogen (see Table 5). These unanticipated results lead us to caution the sole use of ellipsometric measurements to evaluate the antifouling properties of interfaces.

| SAM        | protein loading—ng/cm$^2$ |
|------------|---------------------------|
|            | protamine | lysozyme | BSA  | fibrinogen |
| C16SH      | 392 ± 23  | 541 ± 89 | 789 ± 64 | 792 ± 10 |
| F8H8SH     | 369 ± 150 | 671 ± 31 | 618 ± 76 | 793 ± 54 |
| F8H10-C12  | 253 ± 69  | 406 ± 64 | 410 ± 53 | 571 ± 50 |
| F8H10-C18  | 353 ± 25  | 410 ± 21 | 458 ± 19 | 565 ± 61 |

Values were calculated from QCM data obtained as described in the Supporting Information.

Table 5. Change in Ellipsometric Thickness Values of the SAMs after Exposure to 1 mg/mL of Protein Solution in PBS

| SAM        | thickness (Å) |
|------------|---------------|
|            | protamine | lysozyme | BSA  | fibrinogen |
| C16SH      | 3 ± 1      | 21 ± 4   | 21 ± 6 | 37 ± 3    |
| F8H8SH     | 4 ± 2      | 23 ± 3   | 17 ± 1 | 45 ± 11   |
| F8H10-C12  | 7 ± 1      | 19 ± 1   | 21 ± 2 | 27 ± 2    |
| F8H10-C18  | 3 ± 2      | 22 ± 6   | 16 ± 5 | 30 ± 5    |

■ CONCLUSIONS

The protein-resistant properties of SAMs generated from partially fluorinated spiroalkanedithiols, F8H10-C12 and F8H10-C18, and their monodentate analogues C16SH and F8H8SH were measured using SPR, QCM, and ellipsometry. Proteins having a wide range of physical properties were used to provide a holistic understanding of protein resistance on the SAMs. The studies found that biofilms are less prone to form on the mixed hydrocarbon/fluorocarbon-terminated bidentate SAMs derived from F8H10-C12 and F8H10-C18 than on the single-component monodentate SAMs derived from C16SH and F8H8SH. The enhanced protein resistance observed for the F8H10-C12 SAMs was attributed to the extension of the hydrophobic helical F8 fluorinated chains above the underlying densely packed alkyl chains, thereby exposing greater numbers of antiadhesive interfacial CF$_2$ groups per unit area than in the F8H8SH SAMs. Similarly, the enhanced protein resistance observed for the F8H10-C18 SAMs was attributed to unfavorable interactions between the two phase-incompatible hydrocarbon and fluorocarbon tail groups which give rise to a loosely packed mixture of “conflicted” chains that expose both CH$_2$ and CF$_2$ groups—mixtures that are not found in nature and can plausibly lead to diminished protein adsorption.

The SPR and QCM data provided no convincing evidence that the size/molecular weight of the proteins, the net charge of the proteins, or the surface energies of the SAMs influence protein adsorption in a systematic manner, these results are consistent with a model in which the relative degree of protein adsorption on these low surface energy interfaces is predominantly influenced by the sticky nature of the respective proteins.

The studies also found that the SPR and QCM measurements in most cases were able to distinguish the adsorption trends for the SAMs and proteins examined, but the ellipsometric thickness measurements were less discriminating. The experiments presented herein encompass typical standards for the evaluation of biofilm formation in anticipation of future applications of these model surface coatings and polymeric analogues to follow. Potential substrates include systems ranging from medical implants to oil pipelines and marine-based machines and structures.

■ ASSOCIATED CONTENT

# Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.abm.0c01409.

Details of the materials and experimental methods, ellipsometric thickness values for the untreated SAMs, and details of the surface energy calculations including numerical values provided (PDF)

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Notes
The authors declare no competing financial interest.

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