**Short Communication**

**Thermal analysis, Raman spectroscopy and complex impedance analysis of Cu$^{2+}$-doped KDP**

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**Abstract** Raman spectroscopy and differential thermal analysis (DTA) and thermogravimetric analysis have been carried out on Cu-doped KH$_2$PO$_4$ (Cu-KDP). X-ray diffraction powder data reveal that the structure of the KDP crystal does not change with the additive Cu$^{2+}$ ion. DTA analysis and Raman study of Cu-KDP as a function of temperature reveal that this compound undergoes two phase transitions at about $T_{tr} = 453$ and 473 K. The electrical conductivity measurements on polycrystalline pellet of Cu-KDP (5) are performed from room temperature (RT) up to 495 K. Only one phase transition is observed at 470 K. The activation energy in the migration is 0.42 eV in the temperature range from RT to 470 K. For temperature above 470 K, the activation energy of the superprotonic phase is 1.87 eV.

**Keywords** Phase transformation · Raman spectroscopy · Dielectric properties

**Introduction**

The KDP family compounds (MH$_2$PO$_4$; M = K, Cs, Rb, NH$_4$) are interesting for many reasons [1–6]. In particular, CsH$_2$PO$_4$ (CDP) as a fuel cell electrolyte at temperatures so-called superprotonic behaviour present an abrupt, several-order-of-magnitude jump in its proton conductivity upon heating above the temperature $T_{tr} = 508$ K [7]. It is demonstrated that the above-mentioned proton conductivity enhancement is associated with a polymorphic phase transition from its room temperature monoclinic (P2$_1$/m) phase to a high-temperature dynamically disordered cubic (Pm-3m) CDP modification [8]. Interestingly, the RbH$_2$PO$_4$ (RDP) compound also exhibits a superprotonic transition at 566 K, although at room temperature RDP is not monoclinic as CDP, but tetragonal (I 4 2d). Using synchrotron X-ray studies, Botez et al. [8] demonstrated that heating RDP towards its superprotonic transition leads to an intermediate temperature ($T_{tr} = 383$ K) change of the RDP tetragonal phase into a monoclinic modification, isomorphic (crystallographically identical) to the monoclinic CDP phase. There is another phosphate, NH$_4$H$_2$PO$_4$, which crystallizes at room temperature in tetragonal space group I 4 2d and upon heating does not show an abrupt enhancement of its proton conductivity [9].

Much less is understood about the structural, chemical and physical property changes that occur in KH$_2$PO$_4$ (KDP) upon heating from room temperature toward its melting point. Thermal events observed around $T_{tr} = 458$ K, for example, have been attributed by some authors to a polymorphic phase transition at an intermediate-temperature KDP modification [10], while others have claimed that the behaviour at $T_{tr}$ was in fact due to chemical changes, such as dehydration and onset of partial polymerization of the room temperature of tetragonal KDP phase [11, 12].

Previous investigations of the structural changes of KDP have been devoted, in the past few years, to understand the exact nature of this transition in KDP, which remained a controversial subject. Using X-ray diffraction measurements, Itoh et al. [13] concluded that on heating through 460 K, the crystal system changed from tetragonal to monoclinic, space group P2$_1$ or P2$_1$/m. The lattice parameters at 468 K were given as $a = 7.47$ Å, $b = 7.33$ Å, $c = 14.49$ Å, $\alpha = \beta = 90^\circ$ and $\gamma = 92.2^\circ$. The spots in the Weissenberg photograph at 468 K were only consistently indexed by assuming a twin structure in the monoclinic phase appearing above 460 K. The complete assignment of the reflections was difficult to do, and until now detailed
structural analyses including atomic coordinates have not been done [11]. Moreover, it has been reported that the monoclinic phase of KDP is metastable at temperatures below Ttr and that it reverts to the stable tetragonal phase after being kept for some days in air at room temperature [10, 12–14]. Very recently, temperature-resolved synchrotron X-ray studies have demonstrated that heating KDP towards its superprotonic transition leads to an intermediate temperature (Ttr = 463 K), where the KDP tetragonal phase changes into a monoclinic phase, the same as both isomorphic monoclinic RDP and CDP [15]. Moreover, they observed that monoclinic KDP is stable up to 508 K and indicated that a monoclinic (P21/m)–cubic (Pm3m) transition upon further heating similar to the one responsible for the superprotonic behaviour of CDP and RDP is not precluded. They concluded that the reported lack of superprotonic behaviour in KDP [16] was most likely due to ion size effects and not due to crystal structure considerations.

In view of these results, it appears that further investigations aimed to clarify the structural changes undergone by KDP upon heating above Ttr are worth carrying out.

It is well known that the presence of a small amount of impurities can considerably influence the growth habit, optical properties and dielectric properties of KDP crystals [17–21].

Due to various physicochemical properties, impurities can be selectively incorporated at the surface of the crystal layer and into the kink sites. The presence of metallic impurities may form both isolated and interstitial defect centres [22].

This work presents the evolution of the Raman spectra of single crystal of pure KDP and Cu2+-doped KDP (Cu-KDP) when the temperature is raised from 303 to 503 K. Raman scattering method has been successfully used to reveal structural information about a wide variety of inorganic compounds: phosphate solid acids [23, 24] and sulfate [18, 25, 26]. In particular, Raman spectroscopy is used to demonstrate the effect of heating on the partial decomposition of KDP [25, 27]. Thus, this method is suitable for the present investigation, which presents also the influence of bivalent Cu2+ impurity ions on the electrical properties.

### Table 1

| Concentration of samples (mg/g) |
|---------------------------------|
| Cu-KDP (1)                      | 0.57 |
| Cu-KDP (2)                      | 0.75 |
| Cu-KDP (3)                      | 1.21 |
| Cu-KDP (4)                      | 2.49 |
| Cu-KDP (5)                      | 3.02 |

**Fig. 1** The XRD patterns of Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5)

**Fig. 2** The Pawley refinement of the lattice parameters of Cu-KDP (5). The point symbols represent the observed diffraction pattern, the solid lines represent the calculated pattern and the curves at the bottom of the figure represent the difference between the observed and calculated patterns. The short vertical lines mark the positions of possible Bragg reflections.

**Fig. 3** Thermograms of DSC and TGA for Cu-KDP in the room temperature –573 K range
Experimental details

Five samples, Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5), were prepared. Each sample was prepared by slow evaporation of 20 ml of a saturated aqueous solution of KDP containing, respectively, 0.5, 0.75, 1, 1.5 and 2 g of Cu (OH)$_2$. The concentration is determined by atomic absorption analysis, with a weight error of 2% (Table 1).

Figure 1 compares the XRD patterns of Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5) as well as from KDP references. The characterization of compound is carried out from X-ray diffraction powder data. The intensities of the diffractograms are collected by using a diffractometer Bruker-AXS, type D8 with CuK$_\alpha$ radiation ($\lambda$(K$_\alpha$$_1$) = 1.54060 Å, $\lambda$(K$_\alpha$$_2$) = 1.54443 Å). Diffraction intensities are measured between 10° and 60°, with a 2θ step of 0.02° for 2 s per point. The data are collected at room temperature. The unit cell parameters calculated by using the patterns matching the routine of the FULPROOF program [28] are $a$ = 7.45 Å and $b$ = 6.974 Å. Figure 2 shows the X-ray diffraction (XRD) pattern of Cu-KDP (5). All the reflection peaks of the XRD pattern of the sample are indexed in a single-phase tetragonal KDP structure with I$\bar{4}$2d space group [29].

Thermogravimetric (TGA) and differential thermal analysis (DTA) are performed on polycrystalline with a TGA/DTA Q600 STD TA Instruments apparatus (sample with Pt crucibles, collected by using a diffractometer Bruker-AXS, type D8 with CuK$_\alpha$ radiation ($\lambda$(K$_\alpha$) = 1.54060 Å, $\lambda$(K$_\alpha$) = 1.54443 Å). Diffraction intensities are measured between 10° and 60°, with a 2θ step of 0.02° for 2 s per point. The data are collected at room temperature. The unit cell parameters calculated by using the patterns matching the routine of the FULPROOF program [28] are $a$ = 7.45 Å and $b$ = 6.974 Å. Figure 2 shows the X-ray diffraction (XRD) pattern of Cu-KDP (5). All the reflection peaks of the XRD pattern of the sample are indexed in a single-phase tetragonal KDP structure with I$\bar{4}$2d space group [29].

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Table 2  Raman spectra of pure KDP, Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5)

| Pure KDP | Cu-KDP (1) | Cu-KDP (2) | Cu-KDP (2) | Cu-KDP (2) | Cu-KDP (2) | Assignment |
|----------|------------|------------|------------|------------|------------|------------|
| 116m     | 115m       | 115m       | 118m       | 118m       | 118m       | Lattice vibrations |
| 154m     | 152m       | 154m       | 159m       | 157m       | 160m       |              |
| 187w     | 187w       | 187w       | 188w       | 188w       | 189w       |              |
| 191w     | 191w       | 191w       | 193w       | 193w       | 193w       | v (O–H≡O)  |
| 235m     | 234m       | 235m       | 253m       | 253m       | 253m       | Cu–O        |
| 358m     | 357m       | 357m       | 352m       | 355m       | 354m       | $\delta_1$(PO$_2$, P(OH)$_2$) |
| 391w     | 390w       | 389w       | 388w       | 388w       | 388w       | $\gamma_2$(OH)$_2$ |
| 474m     | 469m       | 473m       | 475w       | 473m       | 473m       | $\delta_1$(PO$_2$, P(OH)$_2$) |
| 536w     | 529w       | 530w       | 532w       | 533w       | 533w       | $\gamma_2$(OH)$_2$ |
| 565m     | 563m       | 563w       | 561w       | 561w       | 561w       | $\delta_2$(PO$_2$, P(OH)$_2$) |
| 916vs    | 916vs      | 916vs      | 914vs      | 914vs      | 914vs      | $\nu_1$(OH)$_2$ |
| 1,817m   | 1,810m     | 1,815m     | 1,759w     | 1,753m     | 1,753m     | Band C      |
| 2,370m   | 2,369m     | 2,368m     | 2,373m     | 2,379m     | 2,383m     | $\nu_2$(OH)$_2$ (band B) |
| 2,728m   | 2,719m     | 2,720m     | 2,829m     | 2,831m     | 2,830m     | $\nu_3$(OH)$_2$ (band A) |

vs very strong, s strong, m medium, w weak
Al₂O₃ as a reference). Thermograms are collected on 28.65-mg samples in the RT–573 K range with a heating rate of 10 K/min under a dry nitrogen flow rate of 100 ml/min.

Raman spectra are measured with a LABRAMHR 800 triple monochromator. The slit widths are set to maintain a resolution of approximately 3 cm⁻¹. The excitation light is 632.81 nm wavelength of He–Ne (20 mV) laser. Measurements are carried out using a microscope in an open furnace (under air, at ambient pressure) heating from 303 to 513 K. The heating rate is 10 K/min, waiting for 1 min after stabilization of temperature, and collecting time for each spectrum is 10 s.

The infrared absorption spectrum was recorded using a pellet of sample, which was prepared by mixing 1 mg sample in a total weight (samples+KBr) of 200 mg. A Perkin-Elmer FT-IR

### Table 3  Infrared spectra of pure KDP, Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5)

| Pure KDP | Cu-KDP (1) | Cu-KDP (2) | Cu-KDP (2) | Cu-KDP (2) | Cu-KDP (2) |
|----------|------------|------------|------------|------------|------------|
| 494m     | 492m       | 490m       | 466m       | 466m       | 463m       |
| 549m     | 546m       | 544m       | 569m       | 568m       | 568m       |
| 829m     | 829m       | 832m       | 886m       | 886m       | 886m       |
| 934m     | 928m       | 926m       | 998m       | 996m       | 996m       |
| 1,083w   | 1,079w     | 1,071w     | 1,108w     | 1,106w     | 1,106w     |
| 1,151w   | 1,149w     | 1,134w     | 1,169w     | 1,168w     | 1,168w     |
| 1,288w   | 1,284w     | 1,281w     | 1,329w     | 1,326w     | 1,326w     |
| 1,635s   | 1,631s     | 1,629s     | 1,683s     | 1,681s     | 1,681s     |
| 2,045w   | 2,038w     | 2,032w     | –          | –          | –          |
| 2,293vw  | 2,288vw    | 2,285vw    | –          | –          | –          |
| 2,343vw  | 2,339vw    | 2,337vw    | –          | –          | –          |
| 2,386vw  | 2,376vw    | 2,373vw    | 2,381m     | 2,383m     | 2,379m     |
| 3,268s   | 3,264s     | 3,261s     | –          | –          | –          |
| 3,345w   | 3,430w     | 3,425w     | –          | –          | –          |
| 3,516w   | 3,512w     | 3,513w     | –          | –          | –          |
| 3,600m   | 3,593m     | 3,592m     | –          | –          | –          |

vs very strong, s strong, m medium, w weak

Al₂O₃ as a reference). Thermograms are collected on 28.65-mg samples in the RT–573 K range with a heating rate of 10 K/min under a dry nitrogen flow rate of 100 ml/min.

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The infrared absorption spectrum was recorded using a pellet of sample, which was prepared by mixing 1 mg sample in a total weight (samples+KBr) of 200 mg. A Perkin-Elmer FT-IR

### Fig. 6  Correlation scheme of internal vibration of the (H₂PO₄)⁻ in tetragonal Cu-KDP

| Free (PO₄)²⁻ | Free (H₂PO₄)⁻ | The frequencies calculated | Site symmetry | Factor group |
|--------------|--------------|---------------------------|---------------|--------------|
| T_d          | C₁v          | DFT/B3LYP(cm⁻¹) | RHF/MP2(cm⁻¹) | C_d          | D₄d          |
| v₁ (938 cm⁻¹) (A₁) | v₁e (POH₂) | 732 | 756 | (A₁) | 5A₁ (R) | 4A₁ (I) | 5B₁ (R) | 4B₁ (I) | 9E (R, IR) |
| v₂ (382 cm⁻¹) (E) | δ₁ [PO₂, P(OH₂)] | 391 | 393 | (A₁) | 5A₁ (R) | 4A₁ (I) | 5B₁ (R) | 4B₁ (I) | 9E (R, IR) |
| v₃ (1033 cm⁻¹) (B₂) | v₁ (PO₂) | 1076 | 1060 | (B₂) | 4B₁ (I) | 9E (R, IR) |
| v₄ (759 cm⁻¹) (F') | δ₂ [PO₂, P(OH₂)] | 419 | 428 | (B₂) | 4B₁ (I) | 9E (R, IR) |

δ: in plane bending; γ: out-of-plane bending; e: rocking; sc: scissors; w: wagging and t: twisting; v: stretching; s: symmetrical, a: antisymmetrical; R: Raman active, IR: IR active and I: Inactive
system PC spectrophotometer in the range of 4,000–400 cm$^{-1}$ using 30 scans with 4-cm$^{-1}$ spectral resolution was used for this purpose. A reference pellet was prepared using 200 mg KBr and has been used to correct the background.

Electrical conductivity measurements were carried out by means of impedance spectroscopy in dehydrated N$_2$ atmosphere in the temperature range from 315 to 495 K. The experiments were done out with ion-blocking sputtered Pt electrodes in a two-probe cell. A frequency response analyser (Solartron 1260) and a dielectric interface (Solartron 1296) were used in the frequency domain from 1 Hz to 10 MHz. Before each measurement, electrochemical system linearity and stability were checked. An AC voltage of 300 mV is used, with a waiting time of 25 min for each 23°C step (thermal equilibration). Impedance diagrams plotted in the Nyquist complex plane were fitted with a series combination of $R_s$ and $R_p/CPE$ elements of the Z-view 3.2c software assigned [30].

Results and discussion

DTA and TGA studies

The thermal measurement (Fig. 3) of Cu-KDP (5) reveals the presence of two high-temperature phase transitions at 462 K (189 °C) and 473 K (200 °C). Thermal gravimetric analysis shows that the transition is not related to decomposition. Significant weight loss occurred from 498 K (225 °C).

Infrared and Raman spectroscopy

The KDP crystal at room temperature may be considered as consisting of K$^+$ and (H$_2$PO$_4$)$^-$ ions belonging to the space group I $\bar{4}$ 2d. The H$_2$PO$_4$ groups together with atoms of K lying between them on axis $z$ (c) create columns shifted one against the other at $c/4$ along the direction $z$. Each PO$_4$ group is linked with four neighbouring PO$_4$ groups by four hydrogen bonds lying almost exactly in the planes perpendicular to the $z$-axis. The local symmetry of the (H$_2$PO$_4$) tetrahedron is $C_2$,
not $S_4$, which is the average site symmetry in the paraelectric phase, as determined by X-ray analysis [5].

At room temperature, the Raman spectrum of Cu-KDP is similar to that of KDP [26, 31]. Figures 4 and 5 show the infrared and Raman spectra of pure KDP, Cu-KDP (1), Cu-KDP (2), Cu-KDP (3), Cu-KDP (4) and Cu-KDP (5). Spectral data and proposed assignment are listed in Tables 2 and 3. The analysis is based on the classical identification of external modes implying the whole lattice network, internal modes of the \((\text{H}_2\text{PO}_4)^-\) tetrahedra and stretching and bending modes of OH bonds. The attribution is done with respect to the KDP attribution taken from [4, 31, 32] and calculated frequencies of free \((\text{H}_2\text{PO}_4)^-\) ions.

To calculate the frequencies of the vibrational modes of the \(\text{H}_2\text{PO}_4\) entity, ab initio RHF/MP2 and DFT/B3LYP electronic structure calculations were carried out using the TZV standard basis set augmented by one diffuse and one polarization functions implemented in the GAMESS program [33]. To identify different vibrational modes, the MOLDEN package [34] was used. Vibrational levels obtained by RHF/MP2 and DFT/B3LYP calculation on free \((\text{H}_2\text{PO}_4)^-\) and in the experimental result are compared with those measured [32] in aqueous solution.

Figure 6 shows the correlation of the free \((\text{H}_2\text{PO}_4)^-\) group vibrations in \(C_{2v}\) internal vibrations in \(C_2\) factor group symmetry through the \(D_{2d}\) one in the crystal.

The spectroscopic characteristics of the free \(\text{H}_2\text{PO}_4\) ion with pseudo-symmetry \(C_{2v}\) can be deduced from the free tetrahedral \(\text{PO}_4\) entity. The two stretching modes $\nu_1$ (~938 cm$^{-1}$) and $\nu_3$ (~1,017 cm$^{-1}$) lead to four stretching modes (two modes $\nu(\text{PO}_2)$, one antisymetrical (B$_1$) at about 1,033 cm$^{-1}$ and one symmetrical (A$_1$) at 1,060 cm$^{-1}$, and two $\nu(\text{OH})$) around 732 and 759 cm$^{-1}$ with symmetry A$_1$ and B$_2$, respectively) (Fig. 6). The bending modes, arising from $\nu_2$ and $\nu_4$, are expected at around 315 and 382 cm$^{-1}$ and at 419, 475 and 484 cm$^{-1}$, respectively. Finally, the three calculated frequencies involving the OH group are 1,093, 1,299 and 3,830 cm$^{-1}$. All of these vibrations theoretically appear in Raman spectra, and only the vibrations with symmetry mode A$_1$, B$_1$, and B$_2$ appear in IR spectra.

As shown in Figs. 4 and 5, the additive has a considerable influence in the Raman and IR spectra. For example, the bands which appear at 3,600 and 3,268 cm$^{-1}$ in pure KDP and assigned to free O–H stretching [19] are absent from Cu-KDP (3). The absence of these peaks supports the adsorption of Cu$^{2+}$ in the surfaces of the crystal. The deviation of IR frequencies for O–H stretching observed at 3,435 and 3,516 cm$^{-1}$ in pure KDP [19] to higher frequencies, respectively, at 3,483 and 3,565 cm$^{-1}$ in Cu-KDP (3) reflects the possible entry of Cu$^{2+}$ cations in the lattice site of the KDP crystal. The increasing frequency of O–H stretching indicates clearly the weakening of the bond strength between oxygen and hydrogen and the interaction of dopant with P–O–H group of KDP. The entry of Cu$^{2+}$ ions into the lattice of the tetragonal KDP crystal can be also confirmed by the variation of Raman and IR frequencies for stretching and bending vibrations $\nu(\text{PO}_2)$, $\nu(\text{OH})$, $\delta(\text{H}_2\text{PO}_4)$ and $\gamma(\text{H}_2\text{PO}_4)$ (Tables 2 and 3).

In the Raman spectra, we observed two new bands centred at 253 and 235 cm$^{-1}$ in the spectrum of Cu-KDP (3). We observed that these two peaks become more intense for Cu-KDP (5). These two bands are characteristic of the Cu–O bond [35].

External modes result from translational and vibrational modes of anions and translational ones of cations. The vibrational
and translational modes of anions can be seen as hydrogen bond stretchings and bendings inside the layers. They are observed below 300 cm⁻¹ for KDP families [36]. According to Som et al., the lines observed at 116 and 154 cm⁻¹ in pure KDP and at 118 and 155 cm⁻¹ in Cu-KDP may be assigned to K–PO₄ translatory vibrations along the c-axis [37]. The 187 cm⁻¹ in Cu-KDP (3) band can be assigned to the H₂PO₄ rotational band. The band observed at 193 cm⁻¹ from Cu-KDP (3) can be attributed to the stretching ν(O–H=O) bridge vibration. This band is observed at about 191 cm⁻¹ in Raman spectra of pure KDP.

The ABC-type broad bands of high-frequency H vibrations have been interpreted as O–H stretching modes in Fermi resonance with combination involving mainly O–H bending vibrations [38] or in terms of strong coupling between fast O–H and slow O=O stretching modes [38]. These spectral characteristics have been observed in a variety of strong hydrogen-bonded solids having O–O distances varying from 2.45 to 2.66 Å [39]. In the present case, the ABC-type bands appear in the Raman spectrum of pure KDP at 2,728, 2,370 and 1,817 cm⁻¹, respectively (Fig. 7). These bands appear at about 2,830, 2,383 and 11,758 cm⁻¹, respectively, in Cu-KDP (5) (Fig. 7). The substantial change in the position of these bands reflects the possible incorporation of Cu²⁺ in the lattice site of KDP crystal.

In conclusion, in a low concentration of copper (Cu-KDP (1) and Cu-KDP (2)), all of the copper is practically substituted in volume, whereas in the high concentrations the copper is absorbed in both volume and the surface.

It is reported that at an increase of the temperature above 453 K, a monoclinic phase is obtained. The tetragonal→monoclinic transition usually occurs at a temperature between 453 and 493 K; the measured value depends strongly on sample preparation [8, 40]. In the monoclinic phase, space group P₂₁/m (C₁₀h), the atomic distribution and lattice parameters at 463 K are obtained (α=7.590 Å, b=6.209 Å, c=4.530 Å, β=107.36°) [8].

In Fig. 8, the increase of temperature from 303 to 503 K shows some modifications in the Raman spectra of Cu-KDP (5) but without any well-known bands due to P₂O₇ entities [40]. These bands might indicate some dehydration (Table 4). Two-phase transitions at 453 and 473 K are related to Cu-KDP (5) (Fig. 9).

The increase of temperature in Cu-KDP (5), between 303 and 503 K, causes the disappearance of lines at low frequency, corresponding to the lattice modes and the band at 193 cm⁻¹, attributed to the stretching O=O bridge vibrations. The appearance in Cu-KDP (5) spectrum of a line at 61 cm⁻¹ and another at 1,175 cm⁻¹ corresponds to ν₁(P=O2) vibrations. Also, there are substantial changes in the position and intensity of some bands corresponding to the internal vibrations of H₃PO₄.

The most important change concerns the Raman line associated to ν₁(P=OH₂) mode. This strong line observed at 914 cm⁻¹ at room temperature splits to two Raman lines near 901 and 960 cm⁻¹ for the temperature between 453 and 463 K and reappears (from 473 K) with a single strong line at 922 cm⁻¹. The frequency of this vibration, ν₁(P=OH₂), estimated by Joost Vande Vondele et al. [32] from the calculation of free H₃PO₄ and in aqueous solution are 795 and 944 cm⁻¹, respectively.

Appreciable changes are also observed for lines at 1,175 cm⁻¹ due to ν₂(P=O₂) vibrations. This line appears between 453 and 473 K and disappears from 483 K.

A plot of the temperature behaviour of some Raman wavenumber is shown in Fig. 10, which unambiguously proves the existence of singularities in the temperature range 303–503 K. From these curves, the temperature of the II→I (a) transition is estimated at 453 and 473 K for I-a→I-b transition.

Table 5 Selected transport properties of Cu-KDP. Activation energies for proton conductivity determined from fit of σ=A/T exp (−Eₐ/KT) over the temperature range stated. Conductivity given for the temperature listed

| Phase | Eₐ (eV) | T range (K) | Sigma (Ω⁻¹ cm⁻¹) | T (K) |
|-------|---------|-------------|------------------|-------|
| I(a)  | 0.42    | 315–490     | 2.2×10⁻⁵         | 315   |
| I(b)  | 1.87    | 450–494     | 3.7×10⁻⁵         | 472   |

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Fig. 11 Complex plane plots of Cu-KDP (5) at various temperatures

Fig. 12 Conductivity plots log (σT)=f (1,000/T) for Cu-KDP (5) compound
Dielectric properties

Some complex impedance diagrams – $Z''$ versus $Z'$ at various temperatures are given in Fig. 10 and show that Cu-KDP (5) follows the Cole–Cole law. The bulk ohmic resistance relative to experimental temperature is the intercept on the real axis of the zero-phase angle extrapolation of the highest frequency curve.

In Fig. 11, the equivalent circuit of the crystal under the ac electric field at lower temperatures can be modelled well as a solution resistance $R_s$ in series with the parallel combination of a polarization resistance $R_P$ with a constant phase angle element (cpe).

The conductivity of Cu-KDP (5) is shown in Arrhenius curve (Fig. 12). Only one transition, at $T_{tr}=450$ K, of the two transitions noted by the calorimetric and Raman studies is apparent in the conductivity results. This transition leads to a sharp increase of the conductivity. The second transition is not observed because it produces a small impact on the transport properties.

The activation energies are obtained as $E_a=0.42$ and 1.87 eV above and below $T_p=450$ K for the ionic hopping of mobile ions (H$^+$) (Table 5). In the KDP material, the activation energies are 0.42 and 0.68 eV for the temperature above and below 453 K, respectively. In KH$_2$AsO$_4$, the activation energies are 0.40 and 0.59 eV for the temperature above and below 453 K [41].

It is noteworthy that the superprotonic Cu-KDP (5) has a rather low activation energy for proton transport (0.42 eV) and a significantly higher conductivity (2.2$\times$10$^{-7}$ $\Omega^{-1}$ cm$^{-1}$) for the polycrystalline pellet compared to the pure KDP (10$^{-10}$ $\Omega^{-1}$ cm$^{-1}$) [5, 42]. It is possible that the higher conductivity is due to the proton defects in Cu-KDP caused by the substitution of K$^+$ by Cu$^{2+}$.

General conclusions

The high temperature phenomena of Cu-KDP are not related to chemical change such as thermal decomposition but related to the physical change of the structural phase transition. Doping KDP with Cu$^{2+}$ significantly modifies its electric properties. The transport properties of protons in such crystals may be treated in terms of proton defects.

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