Decreased bacteria activity on Si$_3$N$_4$ surfaces compared with PEEK or titanium

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Abstract: A significant need exists for orthopedic implants that can intrinsically resist bacterial colonization. In this study, three biomaterials that are used in spinal implants – titanium (Ti), poly-ether-ether-ketone (PEEK), and silicon nitride (Si$_3$N$_4$) – were tested to understand their respective susceptibility to bacterial infection with Staphylococcus epidermidis, Staphylococcus aureus, Pseudomonas aeruginosa, Escherichia coli and Enterococcus. Specifically, the surface chemistry, wettability, and nanostructured topography of respective biomaterials, and the effects on bacterial biofilm formation, colonization, and growth were investigated. Ti and PEEK were received with as-machined surfaces; both materials are hydrophobic, with net negative surface charges. Two surface finishes of Si$_3$N$_4$ were examined: as-fired and polished. In contrast to Ti and PEEK, the surface of Si$_3$N$_4$ is hydrophilic, with a net positive charge. A decreased biofilm formation was found, as well as fewer live bacteria on both the as-fired and polished Si$_3$N$_4$. These differences may reflect differential surface chemistry and surface nanostructure properties between the biomaterials tested. Because protein adsorption on material surfaces affects bacterial adhesion, the adsorption of fibronectin, vitronectin, and laminin on Ti, PEEK, and Si$_3$N$_4$ were also examined. Significantly greater amounts of these proteins adhered to Si$_3$N$_4$ than to Ti or PEEK. The findings of this study suggest that surface properties of biomaterials lead to differential adsorption of physiologic proteins, and that this phenomenon could explain the observed in-vitro differences in bacterial affinity for the respective biomaterials. Intrinsic biomaterial properties as they relate to resistance to bacterial colonization may reflect a novel strategy toward designing future orthopedic implants.

Keywords: silicon nitride, nanostructure, anti-infective, biofilm, protein adsorption

Introduction

In the year 2008, an estimated 413,171 spinal fusion operations, 436,736 primary total hip arthroplasty (THA), and 680,839 primary total knee arthroplasty (TKA) procedures were performed in the United States alone, leading to an approximately US$13 billion market for orthopedic implants. Future demand for such implants is expected to rise. When otherwise well functioning orthopedic implants become colonized with bacteria, significant patient morbidity can follow. The reason is that bacteria adhere to implant surfaces and are resistant to host immune mechanisms and systemic antibiotics. Treatment of such implant-related infections requires extensive repeat surgery, implant extrication, extended duration of systemic antibiotic therapy, bone loss, and substantial cost, suffering, and disability.

Implant-related infections occur in spine surgery between 2.6% and 3.8% of the time. In hip replacement surgery, deep infection occurs at 1.63% at 2-years...
post-operatively, and the figures for knee replacement are about 1.55%. These numbers underestimate the true incidence of infection because of difficulties related to clinical diagnoses, and a paucity of credible data from epidemiological surveys. Mortality rates from complications related to deep prosthetic infections in THA and TKA are reportedly between 2.7% and 18%, and infection is the most common reason for repeat surgery after otherwise successful prosthetic joint replacement. The incidence of infection in THA and TKA may be increasing, probably due to improved detection and also because of evolving antibiotic-resistant bacteria strains. Treatment costs related to infected orthopedic implants can range from 1.52 to 1.76 times the cost of the original surgery. While orthopedic implants are deemed expensive, their costs are easily overwhelmed by the overall treatment cost of implant-related infections. Therefore, there is considerable interest in reducing the risk of infections related to orthopedic implants.

In addition to total joint replacement and spinal devices, orthopedic implants are also manufactured as screws, plates, and percutaneously implanted pins to treat fractures; bacterial contamination of such can lead to poor bone healing. Unless acute implant infections can be overcome expeditiously, adjacent tissues become colonized with bacteria, and the risk of chronic bone infection (osteomyelitis) increases. Osteomyelitis can complicate fracture treatment with metal external fixator devices up to 4% of the time. Ultimately, orthopedic infections lead to implant loosening, nonhealing of fractures, and device failures.

Bacteria present in the bloodstream and body tissues can usually be cleared by host immune mechanisms and antibiotic treatment. However, once bacteria colonize implant surfaces by expressing a biofilm layer, they become relatively impervious to such nonsurgical measures. Biofilm production is accompanied by changes in gene expression and growth rates such that host immune mechanisms are rendered ineffective, leading to a chronically infected environment around the implant. A related clinical concern pertains to the development of antibiotic-resistant strains of bacteria, including S. aureus and S. epidermidis, which have been well documented in the clinical orthopedic setting. Any material science-derived strategy that discourages bacterial colonization of implant surfaces will therefore be of value in reducing the need for systemic antibiotic therapy and surgical removal of infected implants.

Changing implant surface texture from a micron-sized to a nanometer-sized topography affects osteoblast adhesion and subsequent cellular function, while decreasing competitive fibroblast activity. Nanoroughened titanium (Ti) made by electron-beam evaporation and nanotubular and nanotextured Ti created by anodization can enhance osteoblast adhesion and function (such as alkaline phosphatase synthesis, calcium deposition, and collagen secretion) when compared with micron nanosmooth control surfaces. Increased protein adsorption on nanotextured Ti surfaces is correlated with improved osteoblast function. The authors have previously shown that by selectively engineering the surface topography of a biomaterial, bacteria adhesion can be decreased.

Because silicon nitride (Si$_3$N$_4$) is used in spinal reconstructive surgery today, the adhesion of multiple bacteria species onto polished and nanostructured versions of this biomaterial was examined, using two other orthopedic biomaterials (Ti and poly-ether-ether-ketone [PEEK]) as controls. The null hypothesis was that bacterial adhesion and surface protein adsorption (fibronectin, vitronectin, and laminin) would not differ between these materials. Gram-positive S. epidermidis and S. aureus, and Gram-negative P. aeruginosa, E. coli, and Enterococcus were tested because these strains are commonly implicated in orthopedic implant infections.

**Materials and methods**

**Biomaterials**

Si$_3$N$_4$ was supplied by Amedica Corporation (Salt Lake City, UT) in two surface morphologies; with an “as-fired” surface that has nanostructured features, and a smooth polished surface, respectively. Biomedical grade 4 titanium (Fisher Scientific, Continental Steel and Tube Co, Fort Lauderdale, FL) and PEEK Optima® (Invibio, Lancashire, United Kingdom) were obtained with typical machined surfaces. The surfaces of these materials were characterized for morphology and roughness using scanning electron microscopy (SEM) (using an LEO 1530 VP FE-4800 field-emission scanning electron microscope [Zeiss, Peabody, MA]). Specimens of 1 cm × 1 cm dimensions first underwent sessile water-drop tests to assess material wetting characteristics using a KRÜSS easy drop contact angle instrument connected to the drop shape analysis program (version 1.8) (KRÜSS GmbH, Hamburg, Germany) in accordance with an ASTM standard (D7334-08). Prior to bacterial exposure, all samples underwent sterilization with ultra violet light exposure for 24 hours on all sides, and roughness characterization with standard SEM.

**Bacteria studies**

Bacteria were inoculated (10$^8$) onto the material surfaces for 4, 24, 48, and 72 hours. S. epidermidis, P. aeruginosa,
S. aureus, E. coli, and Enterococcus were obtained from American Type Culture Collection (Manassas, VA) (strains 35984, 25668, 25923, 26, and 6569, respectively). The dry pellet was rehydrated in 6 mL of Luria broth (LB) consisting of 10 g tryptone, 5 g yeast extract, and 5 g NaCl per liter double-distilled water with the pH adjusted to 7.4 (Sigma-Aldrich, St Louis, MO). Fetal bovine serum (FBS) (HyClone; Thermo Scientific, Waltham, MA) at 10% concentration was added to the LB, and the bacteria solution was agitated under standard cell conditions (5% CO₂/95% humidified air at 37°C) for 24 hours until the stationary phase was reached. The second passage of bacteria was diluted at a ratio of 1:200 into fresh LB supplemented with 10% FBS and incubated until it reached the stationary phase. The second passage was then frozen in one part LB and 10% FBS and one part glycerol (Sigma-Aldrich) and stored at −18°C. All experiments were conducted from this frozen stock. One day before bacterial seeding for experiments, a sterile 10 µL loop was used to withdraw bacteria from the frozen stock and to inoculate a centrifuge tube with 3 mL of fresh LB supplemented with 10% FBS.

Bacterial function was determined by crystal violet staining and a live/dead assay (Molecular Probes®, Life Technologies, Carlsbad, CA) using previously described methods.³⁵ For the live/dead assay, at the end of the prescribed time period, substrates were rinsed twice with Tris-buffered saline (TBS) comprised of 42 mM Tris-HCl, 8 mM Tris base, and 0.15 M NaCl (Sigma-Aldrich) and then incubated for 15 minutes with the BacLight™ Live/Dead solution (Life Technologies) dissolved in TBS at the concentration recommended by the manufacturer. Substrates were then rinsed twice with TBS and placed into a 50% glycerol solution in TBS prior to imaging. Bacteria were visualized and counted in situ using a Leica DM5500 B fluorescence microscope with image analysis software captured using a Retiga™ 4000R camera (QImaging, Surrey, BC, Canada).

Protein adsorption
Standard fibronectin, vitronectin, and laminin enzyme-linked immunosorbent assays (ELISA) were performed after soaking material samples in bacterial media (LB supplemented with 10% FBS) used above for 20 minutes, 1 hour, and 4 hours, as previously described.³⁵ Substrates were placed in a standard 24-well culture plate and immersed in 1 mL of LB supplemented with and without 10% FBS for 20 minutes, 1 hour, and 4 hours at 37°C in 5% CO₂/95% humidified air. After rinsing in phosphate-buffered saline (PBS), areas that did not absorb proteins were blocked and incubated for 1 hour in bovine serum albumin, BSA (2 wt% in PBS, Sigma-Aldrich). Substrates were again rinsed twice with PBS before either fibronectin, vitronectin, or laminin were directly linked respectively with primary rabbit anti-bovine fibronectin, anti-vitronectin, or anti-laminin (EMD Millipore, Billerica, MA) at a concentration of 6 µg/mL in 1% BSA for 1 hour at 37°C in 5% CO₂/95% humidified air. After rinsing three times with 0.05% Tween 20® (AkzoNobel, Amsterdam, The Netherlands) for 5 minutes with each rinse, the samples were further incubated for 1 hour with a secondary goat anti-rabbit conjugated with horseradish peroxidase (HRP) (Bio-Rad Laboratories, Hercules, CA) at a concentration of 10 µg/mL in 1% BSA. Following triple rinsing with 0.05% Tween 20 for 5 minutes per rinse, surface-adsorbed protein was measured with an ABTS substrate kit (Vector Laboratories, Burlingame, CA) that reacted only with the HRP. Light absorbance was measured at 405 nm on a spectrophotometer and analyzed with computer software. The average absorbance was subtracted by the average absorbance obtained from the negative controls soaked in LB with no FBS. ELISA was performed in duplicate and repeated three times per substrate.

Statistical analysis
Each experiment was done in triplicate with new bacteria, media, and samples. Time series data for biofilm formation, bacteria colonization, and growth were curve-fit to either linear or exponential functions. Hypothesis testing was completed using regression analysis and confidence intervals in accordance with techniques previously described.³⁷ Paired comparison t-tests were used to assess significance for the protein adsorption studies; a P-value of <0.05 was deemed statistically significant.

Results
Comparison of materials
Relevant properties of the three biomaterials tested are compared in Table 1. Si₃N₄ has a protective surface layer that is composed of charged SiNH₂⁺, SiOH⁻, SiO⁻, and neutral SiNH₂ and SiOH groups.³⁸ The presence of the amine groups leads to a high isoelectric point (between 8 and 9) and an overall net positive surface charge.³⁸,³⁹ Ti has a native oxide layer (TiOₓ) which has an isoelectric point of about 4.5 and therefore a negative surface charge at pH 7.⁴⁰ PEEK surfaces have polymeric chains ending in —OH groups, that yield an isoelectric point of about 4.5 and a negative surface charge.⁴¹ Si₃N₄ has the lowest wetting angle and greater hydrophilicity when compared with PEEK or Ti (Table 1).
Table 1 Comparative properties of medical grades of Si$_3$N$_4$, ASTM grade 4 Ti, and PEEK

| Property                        | Units | Si$_3$N$_4$ | Ti-ASTM grade 4 | PEEK optima$^a$ |
|--------------------------------|-------|-------------|-----------------|-----------------|
| Composition                    | NA    | Si$_3$N$_4$, Y$_2$O$_3$, Al$_2$O$_3$ | Chemically pure | Chemically pure |
| Surface composition            | NA    | SiNH$_2$ and SiOH | TiO$_2$ layer | —OH groups     |
| Surface roughness (AFM)        | nm    | 10.1$^*$, 25.3$^{**}$ | 3.06           | 1               |
| Isoelectric point              | NA    | 9           | –4.5           | –4.5            |
| Surface charge at pH = 7       | NA    | Positive    | Negative        | Negative        |
| Sessile water drop wetting angle| Degrees | 39         | 76              | 95              |

Notes: $^a$Polished surface; $^{**}$as-fired surface.

SEM images of biomaterial surfaces are illustrated in Figures 1 and 2. Of the materials tested, PEEK and Ti had machined surfaces, while Si$_3$N$_4$ was in as-fired and polished formulations. SEM images of PEEK and Ti show a micron-rough surface typical of machined materials. The as-fired Si$_3$N$_4$ had nanostructured surface features with a larger total surface area, as shown in the high magnification images (Figure 2). The as-fired Si$_3$N$_4$ surface morphology reflects the natural condition of this material subsequent to densification by sintering and hot-isostatic pressing.$^{42}$ The surface is composed of randomly oriented acicular protruding grains typically less than 1 µm in cross-section, yielding a unique nanotexture (Figure 2B). Note that individual hexagonal grains of Si$_3$N$_4$ have definitive linear facets (in cross-section) and sharp corners at the termination of the acicular grains, which are typically less than 100 nm. These unique features may play a role in interaction with bacteria, resulting in their lysis. This is contrasted to the highly polished Si$_3$N$_4$ surface of Figure 2A in which the acicular asperities were removed.

Surface roughness measurements derived from atomic force microscopy (AFM) measurements are summarized in Table 1. Results were 25 nm, 10 nm, 3 nm, and 1 nm for the as-fired Si$_3$N$_4$, polished Si$_3$N$_4$, Ti, and PEEK, respectively. These AFM surface roughness values differ from those that could be obtained using contact profilometry because the areal sampling size of AFM is small compared with contact profilometry. It is obvious from a comparison of the features in Figure 1 that the micron scale roughness of Si$_3$N$_4$ is similar to that of Ti and PEEK, although there are notable topographical differences. However, the AFM measurements were made over fractions of nanometers. In effect, AFM is measuring the nanoroughness of the materials inbetween micron-sized surface features. These results show that as-fired Si$_3$N$_4$, polished Si$_3$N$_4$, Ti, and PEEK have markedly different surface chemistries and topographies that could affect bacterial and protein adhesion.

![Figure 1](https://example.com/figure1.png)

*Figure 1* Scanning electron microscopy surface microstructures of PEEK Optima$^a$, Ti, and Si$_3$N$_4$: (A1) PEEK 1000×; (A2) PEEK 5000×; (B1) Ti 1000×; (B2) Ti 5000×; (C1) Si$_3$N$_4$ 1000×; (C2) Si$_3$N$_4$ 5000×.

Abbreviation: PEEK, poly-ether-ether-ketone.
Bacteria studies

Figure 3 shows biofilm formation for each bacterial species, and Figure 4 shows corresponding live bacteria counts at each time interval. Except for Enterococcus, trends in biofilm formation were similar for all bacteria. Exponential growth of biofilm was noted on PEEK when exposed to S. epidermidis, S. aureus, P. aeruginosa, and E. coli, whereas linear growth was observed for Si$_3$N$_4$ and Ti. Independent of time points, the lowest biofilm formation occurred on the as-fired Si$_3$N$_4$, followed by polished Si$_3$N$_4$. These differences were statistically significant ($P < 0.05$) for bacterial exposure times > 24 hours. Also, with the exception of Enterococcus, biofilm formation was significantly lower ($P < 0.05$) on Ti compared with PEEK, for time periods > 48 hours. PEEK demonstrated the highest biofilm affinity, with values that were 5–16 times greater than those for as-fired Si$_3$N$_4$, 1.5–10 times more than for polished Si$_3$N$_4$, and between 1 and 6.7 times higher than for Ti.

Live bacteria counts were the highest for PEEK at all time periods, followed by Ti, polished Si$_3$N$_4$, and then as-fired Si$_3$N$_4$ (Figure 4). These differences were statistically significant ($P < 0.05$) for all bacterial species tested, and at all time periods > 48 hours. Live bacteria manifested on PEEK were between 8 and 30 times the bacteria found on as-fired Si$_3$N$_4$. These results were particularly remarkable for P. aeruginosa, which is a virulent and difficult microbe to eliminate from polymeric implants.41

To summarize, less bacterial activity was manifested on Si$_3$N$_4$ than on either Ti or PEEK, probably because of the surface chemistry and nanostructure differences between these materials. Substantially lower loads of S. epidermidis, S. aureus, P. aeruginosa, E. coli, and Enterococcus bacteria were measured on Si$_3$N$_4$ surfaces than Ti or PEEK, for up to 72 hours of incubation.

Protein adsorption studies

Figure 5 shows the relative absorbance of fibronectin, vitronectin, and laminin at 20-, 60-, and 240-minute intervals. Fibronectin and vitronectin adsorption was significantly greater on Si$_3$N$_4$ (whether polished or as-fired) when compared with PEEK or Ti at all time intervals ($P < 0.01$). As-fired Si$_3$N$_4$ showed a higher affinity for these two proteins than polished Si$_3$N$_4$. Laminin adsorption did not differ significantly between the biomaterials at the 20-minute time interval. At 60 minutes, both Si$_3$N$_4$ surfaces showed significant laminin adsorption ($P < 0.01$) when compared with Ti or PEEK. At 240 minutes, as-fired Si$_3$N$_4$ was the only surface that showed statistically more laminin adsorption ($P < 0.01$) than all other materials.

To summarize, Ti and PEEK proved inferior in terms of protein adsorption when compared with as-fired and polished Si$_3$N$_4$ surfaces. As-fired Si$_3$N$_4$ provided the highest protein adsorption platform among the materials tested. Surface protein adsorption is relevant because previous studies have correlated increased vitronectin and fibronectin adsorption to decreased bacterial activity.35

Discussion

Bacterial infection of orthopedic implants is a complex, multifactorial process that is influenced by bacterial properties, the presence of serum proteins, fluid flow around the implant, implant surface chemistry and morphology, host immune variables, and probably other variables.44 Infection can arise from inadvertent contamination of an implant, contagion from the surgical staff, bacteria arising from the patient’s skin or mucus membranes, unrecognized infection elsewhere in the body; ineffectively applied surgical disinfectants, and sepsis acquired from others.45 Most of these risk factors can be mitigated by appropriate nosocomial hygiene practices, and through the use of perioperative antibiotic prophylaxis.5,6,9,46–48 However, no strategy has proven effective in completely eliminating the risk of infections related to implant surgery. Strategies to discourage bacterial adhesion to implants have included antibacterial surface coatings and treatments, and the development and use of nanostructured surfaces, or a combination of these methods.

Surface coatings and treatments

Surface coatings or surface treatments on implants can resist pathogen adhesion or release chemicals that invade bacterial biofilm. Silver ions have been investigated as a surface antiseptic agent, although the exact mechanism for silver ion toxicity on bacteria is unclear.48 Silver can be incorporated into polymeric or inorganic coatings,35,49 and while its anti-infective properties have been known for generations, it continues to drive patent innovation.50,51 Other
coatings have included pre-loaded formulations of antibiotics like vancomycin or gentamicin, or antiseptic agents like chlorohexidine and chloroxylenol. Antibacterial polymeric functionalized coatings which incorporate hyaluronic acid and chitosan have been investigated thoroughly. Xerogel polymers that have been modified to release nitric oxide are effective in inhibiting the adhesion of S. aureus, S. epidermidis, and E. coli. A limitation of all such coatings is the transient duration of effectiveness, ie, once the coating dissipates, so does any antibacterial effect.

Surface modifications of biomaterials can provide a long-term shield against bacterial infection of implants. Yoshinari et al modified the surface of Ti and studied bacterial adhesion of P. gingivalis and Aggregatibacter actinomycetemcomitans using ion implantation of Ca⁺, N⁺, and F⁻, ion-beam deposition of Ag, Sn, Zn, and Pt, ion plating

Figure 3 Biofilm formation: (A) Staphylococcus epidermidis, (B) S. aureus, (C) Pseudomonas aeruginosa, (D) Escherichia coli, (E) Enterococcus. Note: Dashed lines represent confidence intervals at P < 0.05.

Abbreviations: SN, Si₃N₄; PEEK, poly-ether-ether-ketone.
of TiN and Al₂O₃, and anodic oxidation formation of TiO₂.⁵⁹ Control materials included polished, sand-blasted, and striated Ti. They found a general trend toward higher bacterial adhesion on blasted and Ca⁺ implanted Ti. A follow-up study by these authors also showed that F⁺ implanted surfaces resisted initial bacterial adhesion in the absence of fluorine leaching from the surface; the antibacterial effect probably related to metal-fluorine complexes at the surface.⁶⁰ Katsikogianni et al found otherwise: S. epidermidis activity on polymers with and without fluorine showed that those containing fluorine increased bacteria attachment.⁶¹ Raulio et al also investigated fluorine, and found that by coating stainless steel with fluoropolymers,
it was possible to reduce biofilm formation of several bacteria strains, including *S. epidermidis*.62

Li et al performed a comparative study using glass and metal-oxides applied as thin films.63 These included three uncoated glass surfaces and combinations of Co-Fe-Cr, Ti-Fe-O, SnO2, SnO2-F, SnO2-Sb, Al2O3, and Fe2O3 thin films applied to glass substrates. After measuring material hydrophilicity, zeta potential, and surface energy, the authors tested them against eight strains of bacteria, and found that hydrophobicity and total surface energy (rather than material chemistry) led to increased bacterial adhesion. Hydrophilic surfaces had the fewest number of adherent bacteria, and increasing the concentration of surface ions encouraged binding for both Gram-positive (*Bacillus subtilis*) and Gram-negative (two *P. aeruginosa* strains, three *E. coli* strains, and two *Burkholderia cepacia* strains) bacteria.

Conversion of Ti surfaces to TiO2 via anodic oxidation, electrophoretic deposition, chemical vapor deposition, ion implantation, and plasma spraying has been examined as an anti-infective surface modification strategy, especially when coupled with photocatalytic activation of TiO2.64 Orthopedic implants made of porous Ti are commonly coated with hydroxyapatite to promote osteointegration, but this strategy may lead to increased harboring of *S. aureus* bacteria, more severe infection, and less osteointegration when compared with uncoated Ti.65,66

The results of this present study show that Si3N4 has intrinsic bacteriostatic properties, whether in an as-fired or smooth-surface morphology. This study did not elucidate the precise mechanism of Si3N4 anti-infectivity, but results suggest that material hydrophilicity and surface chemistry may contribute to less biofilm formation (Figure 3) and lower bacteria counts (Figure 4) on Si3N4 when compared with Ti or PEEK. These differences were observed for all bacteria, regardless of Gram-positive or Gram-negative strains. Hydrophilic Si3N4 surfaces were probably less conducive to bacterial adhesion, when compared with hydrophobic surfaces where water displacement is not required for microbial adherence.64 Consistent with this observation, the inert and highly hydrophobic characteristics of PEEK promoted bacterial adhesion and inhibited protein adsorption when compared with Si3N4. Other authors have supported these observations as they relate to PEEK and Ti.67,68

The anti-infectivity of Si3N4 may also relate to surface chemistry, and the authors’ observations are consistent with previous data with polymeric coatings containing chitosan, a material similar to Si3N4 in terms of a net positive charge and the presence of amine groups at the surface.52–56 The interaction of these groups with negatively charged bacteria reportedly leads to membrane disruption and lysis in chitosan-containing polymers.56 However, further research is required to confirm whether or not this is a dominant operative mechanism for Si3N4.
Nanostructured surfaces

Nanotechnology is of interest in enhancing the surface characteristics and improving the performance of orthopedic implants. Increasing micron-level surface roughness of biomedical implants correlates with increased bacterial adhesion and enhanced osteointegration. The competition between bacteria, bone cells, and serum proteins has been characterized as a “race to the surface.” However, surface roughness is only one of several variables that influence bacterial adhesion and protein adsorption on an implant. For instance, without specifically distinguishing changes in surface roughness, surface chemistry, or crystallinity, Truong et al showed that nanoscale roughening of ultrafine grained Ti led to greater attachment of *S. aureus* and *P. aeruginosa*, when compared with smooth surfaces. Conversely, Singh et al found increased bacterial adhesion and biofilm formation on Ti surfaces with Ra (roughness average) values < 20 nm, and increased protein adsorption at Ra values between 16 and 32 nm; further increases in surface roughness were associated with reduced pathogen activity. In contrast, Hilbert et al could not correlate bacterial adhesion, colonization, and growth with changes in surface finish ranging from 0.1 to 0.90 µm, and Flint et al found that bacterial adhesion could not be related to surface roughness at all. Thus, surface roughness alone may not fully explain the differences in bacterial adhesion and protein adsorption.

Anselme et al reviewed the factors that affect the interaction of cells and bacteria with nanostructured surfaces. In addition to surface roughness, detailed surface topography, including the size, shape, orientation, distance, and organization of surface nanofeatures were identified as important variables. Whitehead et al further supported this view when they found higher bacterial counts on substrates that contained 2 µm diameter pits as opposed to those in the range of 0.5 µm. Xu et al compared bacterial adhesion between two polyurethane surfaces of the same chemistry; one was smooth and the other contained oriented protrusions or pillars (400 nm diameter × 650 nm height). They found biofilm formation significantly less for both *S. epidermidis* and *S. aureus* when using the highly textured polyurethane. Dalby et al examined the response of fibroblasts to nanocolumn structures (100 nm diameter × 160–170 nm height) in polymethylmethacrylate (PMMA) in comparison with smooth substrates and demonstrated decreased cell adhesion using the materials with nanocolumns. Bagno et al showed that the number of protruding peaks (ie, peak density) was positively correlated with osteoblast adhesion. Similar topographical effects on bacterial adhesion and protein adsorption were observed by these authors for Ti, when conventional (smooth) surfaces were compared to nanoroughened, nanotextured, and nanotubular variations. The authors also reported decreased in-vitro activity of *S. aureus*, *S. epidermidis*, and *P. aeruginosa* on nanorough surfaces prepared by electron-beam evaporation, but an opposite result for nanotubular and nanorough surfaces prepared via anodization. However, all three of the nanoprepared Ti surfaces demonstrated increased adsorption of fibronectin in comparison with conventional Ti. A subsequent study showed reduced adhesion of macrophages on anodized Ti with nanotextured and nanotubular structures.

Engineered surfaces with topographical features in the nanometer range may affect cell behavior while reducing bacterial adhesion, but these factors alone do not sufficiently predict cellular response. Other variables related to surface energy, surface charge, and chemistry can also inhibit bacterial colonization while enhancing protein adsorption and conformation, leading to improved osteoblast adhesion and tissue growth. Biomaterial surfaces with nanorough features are associated with greater vitronectin and fibronectin adsorption that, in turn, are related to decreased bacterial function on the surface.

The acicular grain structure on the as-fired surface of Si₃N₄ is composed of randomly oriented columnar grains that are typically 200–700 nm in diameter, and that can protrude above the surface up to about 3 µm. This as-fired surface is not only significantly rougher than Ti, PEEK, or polished Si₃N₄ (Table 1) but its nanotopography is fundamentally different, as suggested by AFM surface measurements. Even though the PEEK and titanium samples were as-machined, they had significantly smoother surfaces at the nanolevel than either as-fired or polished Si₃N₄. Consequently, bacterial and protein adherence to the biomaterials differed in the present experiments. While this study did not elucidate the precise mechanisms, it is probable that the observed behavior of as-fired Si₃N₄ is related to nanocolumns, protrusions, and peaks described in previous studies for polyurethane, PMMA, and Ti surfaces.

In summary, the present authors believe it is the totality of silicon nitride’s nano-topography – including its protruding acicular grains with their definitive hexagonal features and sharp grain ends – combined with its surface chemistry (ie, positive surface charge and the presence of amine groups) which provides Si₃N₄ with its unique antibacterial and protein adsorption characteristics.
Conclusion

This study examined the behavior of Ti, PEEK, and Si$_3$N$_4$ when respective materials were exposed to five different bacterial species for up to 72 hours. Polished and as-fired surface formulations of Si$_3$N$_4$ were tested. Decreased biofilm formation and bacterial colonization were demonstrated on both as-fired and polished Si$_3$N$_4$ in comparison with Ti or PEEK, under in-vitro incubation for time periods of up to 72 hours. Si$_3$N$_4$ resisted bacterial proliferation under these conditions, despite the absence of antibiotic pharmaceutical agents. Differential protein adsorption on Ti, PEEK, and Si$_3$N$_4$ surfaces was also demonstrated, such that fibronectin, vitronectin, and laminin adsorbed preferentially onto Si$_3$N$_4$ when compared with Ti or PEEK, for time periods of up to 4 hours. Previous studies have shown that such differences in protein affinity for biomaterial surfaces may explain, at least in part, differences in biomaterial susceptibility to bacterial infection. The unique surface chemistry and nanostructured features of Si$_3$N$_4$ may contribute to the favorable antibacterial properties of this bioceramic material, and support further investigation into the use of Si$_3$N$_4$ as a material platform for orthopedic implants.

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Disclosure

B Sonny Bal is advisory surgeon to Amedica, developer of synthetic silicon nitride for orthopedic applications, and serves on the Board of Directors of Amedica, Salt Lake City, UT. Co-authors Deborah Gorth, Sabrina Puckett, Batur Erkan, Thomas J Webster, and Mohamed N Rahaman have no disclosures for this article.

References

1. Rajaei SS, Bae HW, Kanim LE, Delamarter RB. Spinal fusion in the United States: analysis of trends from 1998 to 2008. Spine. 2012;37(1):67–76.
2. Mendenhall S. Hip and knee implant review. Orthopedic Network News. 2011;22(3):466–469.
3. Collins I, Wilson-MacDonald J, Chami G, et al. The diagnosis and management of infection following instrumented spinal fusion. Eur Spine J. 2008;17(3):445–450.
4. Ong KL, Kurtz SM, Lau E, et al. Prosthetic joint infection risk after total hip arthroplasty in the Medicare population. J Arthroplasty. 2009;24(Suppl 6):105–109.
5. Kurtz SM, Ong KL, et al. Prosthetic joint infection risk after TKA in the Medicare population. Clin Orthop Relat Res. 2010;468:52–56.
6. Matar WY, Jafari SM, Restrepo C, et al. Preventing infection in total joint arthroplasty. J Bone Joint Surg Am. 2010;92 Suppl 2:36–46.
7. Gardner J, Gieoe TJ, Tatan M. Can this prosthesis be saved? Implant salvage attempts in infected primary TKA. Clin Orthop Relat Res. 2011;469(4):970–976.
8. Kurtz SM, Lau E, Schmier J, et al. Infection burden for hip and knee arthroplasty in the United States. J Arthroplasty. 2008;23(7):984–991.
9. Darouiche RO. Treatment of infections associated with surgical implants. N Engl J Med. 2004;350(14):1422–1429.
10. Grisitina AG. Biomaterial-centered infection: microbial adhesion versus tissue integration. Science. 1987;237(4822):1588–1595.
11. Moroni A, Vannini F, Mosca M, Giannini S. State of the art review: techniques to avoid pin loosening and infection in external fixation. J Orthop Trauma. 2002;16(3):189–195.
12. Mahan J, Seligson D, Henry S, Hynes P, Dobbins J. Factors in pin tract infections. Orthopedics. 1991;14(3):305–308.
13. Zlowodzki M, Prakash JS, Aggarwal NK. External fixation of complex femoral shaft fractures. Int Orthop. 2007;31(3):409–413.
14. Green S, Ripley M. Chronic osteomyelitis in pin tracks. J Bone Joint Surg Am. 1984;66(7):1092–1098.
15. Costerton JW, Stewart PS, Greenberg EP. Bacterial biofilms: a common cause of persistent infections. Science. 1999;284(5418):1318–1322.
16. Rimondini L, Fini M, Giardino R. The microbial infection of biomaterials: a challenge for clinicians and researchers. A short review. J Appl Biomater Biomech. 2005;3(1):1–10.
17. Fux C, Costerton J, Stewart P, Stoodley P. Survival strategies of infectious biofilms. Trends Microbiol. 2005;13(1):34–40.
18. Donlan RM. Biofilms and device-associated infections. Emerg Infect Dis. 2001;7(2):277–281.
19. Baquero F. Gram-positive resistance: challenge for the development of new antibiotics. J Antimicrob Chemother. 1997;39 Suppl A:1–6.
20. Campoccia D, Montanaro L, Arciola CR. The significance of infection related to orthopedic devices and issues of antibiotic resistance. Biomaterials. 2006;27(11):2331–2339.
21. Webster TJ, Ejiofor JU. Increased osteoblast adhesion on nanophase metals: Ti, Ti6Al4V, and CoCrMo. Biomaterials. 2004;25(19):4731–4739.
22. Webster TJ, Siegel RW, Bizios R. Osteoblast adhesion on nanophase ceramics. Biomaterials. 1999;20(13):1221–1227.
23. Webster TJ, Ergun C, Doremus RH, Siegel RW, Bizios R. Enhanced functions of osteoblasts on nanophase ceramics. Biomaterials. 2000;21(17):1803–1810.
24. Webster TJ, Ergun C, Doremus RH, Siegel RW, Bizios R. Enhanced osteoclast-like cell functions on nanophase ceramics. Biomaterials. 2001;22(11):1327–1333.
25. Ejiofor J, Webster T. Bone cell adhesion on titanium implants with nanoscale surface features. Int J Powder Metallurgy. 2004;40(2):43–53.
26. Ward BC, Webster TJ. Increased functions of osteoblasts on nanophase metals. Mater Sci Eng C Mater Biol Appl. 2007;27(3):575–578.
27. Miller DC, Vanc RJ, Thapa A, Webster TJ, Haberstroh KM. Comparison of fibroblast and vascular cell adhesion to nano-structured poly(lactic-co-glycolic acid) films. Appl Bionics Biomech. 2005;2(1):1–7.
28. Cohen A, Liu-Synder P, Storey D, Webster TJ. Decreased fibroblast and increased osteoblast functions on ionic plasma deposited nanostructured Ti coatings. Nanoscale Res Lett. 2007;2(8):385–390.
29. Puckett S, Parea R, Webster TJ. Nano rough micron patterned titanium for directing osteoblast morphology and adhesion. Int J Nanomedicine. 2008;3(2):229–241.
30. Yao C, Slamovich EB, Webster TJ. Enhanced osteoblast functions on anodized titanium with nanotube-like structures. *J Biomed Mater Res A*. 2008;85(1):157–166.

31. Woo KM, Chen VJ, Ma PX. Nano-fibrous scaffolding architecture selectively enhances protein adsorption contributing to cell attachment. *J Biomed Mater Res A*. 2003;67(2):531–537.

32. Khanga D, Kim SY, Liu-Snyder P, et al. Enhanced fibroconnectin adsorption on carbon nanotube/poly(carbonate) urethane: Independent role of surface nano-roughness and associated surface energy. *Biomaterials*. 2007;28(32):4756–4768.

33. Webster TJ, Ergun C, Doremus RH, Siegel RW, Bizios R. Specific proteins mediate enhanced osteoblast adhesion on nanophase ceramics. *J Biomed Mater Res*. 2000;51(3):475–483.

34. Colon G, Ward BC, Webster TJ. Increased osteoblast and decreased *Staphylococcus epidermidis* functions on nanophase ZnO and TiO2. *J Biomed Mater Res A*. 2006;78(3):595–604.

35. Puckett SD, Taylor E, Raimondo T, Webster TJ. The relationship between the nanostructure of titanium surfaces and bacterial attachment. *Biomaterials*. 2010;31(4):706–713.

36. ASTM D7334-08 Standard Practice for Surface Wettability of Coatings, Substrates, and Pigments by Advancing Contact Angle Measurement. 2008.

37. Brown AM. A step-by-step guide to non-linear regression analysis. *Int Orthop*. 2011;35(2):253–260.

38. Kim S, Baik S. Hot isostatic pressing of sintered silicon nitride. *J Mater Sci Med*. 2000;11(3):1735–1738.

39. Barton AJ, Sagers RD, Pitt WG. Bacterial adhesion to orthopedic implant polymers. *J Biomed Mater Res*. 1996;30(3):403–410.

40. Katsikogianni M, Missirlis YF. Concise review of mechanisms of the nanostructure of titanium surfaces and bacterial attachment. *Biomaterials*. 2010;31(4):706–713.

41. ASTM D3357-90 Standard Terminology for Surface Characterization of Materials. 2005.

42. Neumann H-G, Prinz C. Method for producing an anti-infective coating on implants. US Patent Application Publication 2012/0024712. February 2, 2012.

43. Zhao L, Chu PK, Zhang Y, Wu Z. Antibacterial coatings on titanium implants. *J Biomed Mater Res B Appl Biomater*. 2009;91(1):470–480.

44. Hamilton V, Yuan R, Rigney D, et al. Bone cell attachment and growth on well-characterized chitosan films. *Polym Int*. 2007;104:641–647.
76. Xu L-C, Siedlecki C. Submicron-textured biomaterial surface reduces staphylococcal bacterial adhesion and biofilm formation. Acta Biomater. 2012;8(1):72–81.

77. Dalby MJ, Riehle MO, Sutherland DS, Agheli H, Curtis ASG. Fibroblast response to a controlled nanoenvironment produced by colloidal lithography. J Biomed Mater Res A. 2004;69(2):314–322.

78. Bagno A, Genovese M, Luchini A, et al. Contact profilometry and correspondence analysis to correlate surface properties and cell adhesion in vitro of uncoated and coated Ti and Ti6Al4V disks. Biomaterials. 2004;25(12):2437–2445.

79. Rajyalakshmi A, Ercan B, Balasubramanian K, Webster TJ. Reduced adhesion of macrophages on anodized titanium with select nanotube surface features. Int J Nanomedicine. 2011;6:1765–1771.

80. Anagnostou F, Debet A, Pavon-Djavid G, et al. Osteoblast functions on functionalized PMMA-based polymers exhibiting Staphylococcus aureus adhesion inhibition. Biomaterials. 2006;27(21):3912–3919.