Multiscale mechanisms of nutritionally induced property variation in spider silks

Sean J. Blamires1*, Madeleine Nobbs1, Penny J. Martens2, I-Min Tso3, Wei-Tsung Chuang4, Chung-Kai Chang4, Hwo-Shuenn Sheu4

1 Evolution & Ecology Research Centre, School of Biological, Earth & Environmental Sciences D26, The University of New South Wales, Sydney, Australia, 2 Graduate School of Biomedical Engineering, Samuels Building F25, The University of New South Wales, Sydney, Australia, 3 Department of Life Science, Tunghai University, Taichung, Taiwan, 4 National Synchrotron Radiation Research Centre, Hsinchu, Taiwan

* s.blamires@unsw.edu.au

Abstract

Variability in spider major ampullate (MA) silk properties at different scales has proven difficult to determine and remains an obstacle to the development of synthetic fibers mimicking MA silk performance. A multitude of techniques may be used to measure multiscale aspects of silk properties. Here we fed five species of Araneoid spider solutions that either contained protein or were protein deprived and performed silk tensile tests, small and wide-angle X-ray scattering (SAXS/WAXS), amino acid composition analyses, and silk gene expression analyses, to resolve persistent questions about how nutrient deprivation induces variations in MA silk mechanical properties across scales. Our analyses found that the properties of each spider’s silk varied differently in response to variations in their protein intake. We found changes in the crystalline and non-crystalline nanostructures to play specific roles in inducing the property variations we found. Across treatment MaSp expression patterns differed in each of the five species. We found that in most species MaSp expression and amino acid composition variations did not conform with our predictions based on a traditional MaSp expression model. In general, changes to the silk’s alanine and proline compositions influenced the alignment of the proteins within the silk’s amorphous region, which influenced silk extensibility and toughness. Variations in structural alignment in the crystalline and non-crystalline regions influenced ultimate strength independent of genetic expression. Our study provides the deepest insights thus far into the mechanisms of how MA silk properties vary from gene expression to nanostructure formations to fiber mechanics. Such knowledge is imperative for promoting the production of synthetic silk fibers.

Introduction

Spider major ampullate (MA) silk is nature’s toughest materials [1]. Accordingly, there is considerable interest in the creation of materials that mimic its performance [2]. Nevertheless, attempts to recombine, amplify and spin spider silk proteins have not produced fibers with properties resembling those of naturally spun silk [3,4]. One reason for the inability to produce
such fibers is that the properties of MA silk are highly variable and the mechanisms inducing this variation have never been delineated from nano to macro scales [5].

Researchers can determine the consequences of gene expression on the functional properties of structural proteins by either switching genes on and off and observing the consequences in the secreted proteins, or by observing the function of the proteins produced in different organisms with varying levels of expression of a particular gene [5]. Indeed, using such approaches has informed us how and why certain genes cause specific ailments in humans and other animals [5,6]. Progress has been recently made into our understanding of spider silk genes and their expression patterns [7–11]. Likewise, significant insights have been gained on silk production, spinning, and its engineering [11–15]. However, because no single study has holistically examined the consequences of gene expression on protein structure and silk functional properties, it is not known why spiders spin silks with such exceptional properties and, more importantly, why silk properties vary so much between and within individual spiders [11,14]. Experimentally switching silk related genes on and off within individual spiders is not yet achievable, so observing the function of silk proteins in different spiders with varying levels of expression of particular genes appears the best way forward.

MA silk properties have traditionally been thought to be the product of the combined expression of genes coding for two proteins (called spidroins); major ampullate spidroin 1, or MaSp1, and major ampullate spidroin 2, or MaSp2 [11,16] (with the genes called MaSp1 and MaSp2). The secondary structures of the spidroins are considered critical for silk performance [17,18]. MaSp1 consists of repeating polyalanine, (GA)_n (GGX)_n and (A)_n amino acid motifs (G = glycine, A = alanine and X = other amino acids). These motifs form cassettes that combine to promote the formation of crystalline β-sheet nanostructures in the assembled fibers [18,19]. The MaSp2 protein on the other hand has been thought to consist of multiple (GPGXX)_n motifs (where P = proline), and predicted to form disordered type II β-turns and similar non-crystalline nanostructures [20]. Collectively, the various nanostructures are thought to combine and provide MA silk with its great strength and extensibility [21]. Since MaSp2 has long been predicted to contain (GPGXX)_n sequences, the proline composition of MA silk was considered a reliable indicator of MaSp2 gene expression [22,23]. The ratio of MaSp1: MaSp2 expression is variable among and between spider species, presumably because the MaSp2 protein is metabolically costly to synthesize so may be differentially expressed. Variation in the ratio of the two spidroins has been traditionally thought to bring about variations in nanostructure formations leading to variations in the mechanics of the spun fibers [11].

The recent mapping of the Nephila clavipes‘ spidroin genome [10] has caused researchers to rethink much assumed knowledge. We now understand that, at least in N. clavipes: (i) individual spiders express multiple versions of the MaSp genes at different loci, potentially leading to several MaSp1 and MaSp2 proteins appearing in the spun silks, (ii) the different spidroin cassettes and motifs are shared and partitioned among the different proteins, and (iii) other spidroins, e.g. AcSps, appear in the major ampullate gland, so can provide additional cassettes and motifs that may form components of MA silk [10,16,24]. For these reasons, MA silk amino acid compositions may not necessarily reflect the MaSp1: MaSp2 expression, possibly explaining why nanostructure formations can vary within individual spider silks independent of MaSp1: MaSp2 expression [25–27].

Any study aspiring to understand how spidroin gene expression influences MA silk nanostructure formation must reliably measure the size, density, orientation, and distances between the crystalline and non-crystalline nanostructures in the silk proteins in addition to genetic expression. Techniques such as small angle X-ray scattering (SAXS), wide angle X-ray scattering (WAXS), nuclear magnetic resonance (NMR), Fourier transform infrared (FTIR), Raman and circular dichroism (CD) spectroscopy, and transmission electron microscopy, have been
used to examine the nanostructures of MA silk’s proteins [28–31]. Of these only synchrotron-derived SAXS and WAXS can explicitly and reliably measure nanoscale variability in silk crystallinity, and the size, density, orientation, and distance between individual crystalline and non-crystalline nanostructures [32,33]. For instance, SAXS derived parameters such as the meridional peak and long period can be used to elucidate nanostructure alignment in the silk’s amorphous and lamellar regions [34–36]. WAXS on the other hand can be used to determine silk crystallinity, and the size, density, and orientation, of crystalline nanostructures by examining the scattering pattern and diffraction angles (θ) at specific high intensity diffraction peaks [28,37,38]. For spider silk the diffraction peaks at the (200), (120) or (002) regions identified on two dimensional WAXS images are of particular interest as they are associated with scattering from crystalline β–sheets [37,39–41]. SAXS and WAXS accordingly are tools appropriate for measuring and classifying silks based on the size, alignment, and distances, between and within crystalline and non-crystalline nanostructures [37,41,42].

X-ray scattering analyses of the MA silks of different spider species has found that the micro-arrangement of silks from spiders in the Araneoidea clade are relatively conserved across the group [41]. The mechanical properties of Araneoid spider MA silks, nonetheless, may vary considerably between species and individuals across environments and loading pre-conditions [43]. For instance, silk spun by spiders walking along a surface is not as stiff as that spun by free falling spiders [43,44]. Much of this variability is considered a consequence of changes in the friction acting on the silk as it exits the valve during spinning since this induces the amorphous region proteins to further self-align [30,43]. The properties of the crystalline nanostructures can nonetheless also vary within individual spider silks across loading conditions. For example, spiders exposed to winds of different strength produce silks differing in nanocrystal density, which affects the ultimate strength of the fibers [25]. Exposing spiders to strong wind induces silk extensibility and ultimate strength to change in the same direction [25,45,46]. Simulations and experiments have shown that variations in glandular pH, salts and shear stress during spinning induce the poly-alanine residues to undergo α-helix → β-sheet phase transitions, which enhances silk strength [47–50]. These types of phase transitions could explain the enhancement of ultimate strength from spiders exposed to high wind. Moreover, it has been shown that the additional enhancement of extensibility in the silks of spiders exposed to strong wind could be a consequence of spinning under a static load, since this induces the amorphous region proteins to move more freely relative to each other [46].

An additional experiment found that the spider Nephila pilipes produces stronger and tougher MA silk when fed a high protein diet than when deprived of protein [27]. Subsequent WAXS analyses found that, like wind exposed spiders [25], changes in silk crystal density explains some of the enhanced strength [27]. Nevertheless, unlike wind exposed spiders, a combination of changes in crystallinity and nanostructural orientation in the amorphous region were also prevalent [27]. In both instances, the silk nanostructures seem to vary independent of MaSp1: MaSp2 expression, although only indirect measures (i.e. amino acid composition) of MaSp expression was made [25,27]. Different mechanisms at different scales seem to be responsible for nutritionally induced spider silk property changes compared to wind induced property changes. However, no study has holistically examined the consequences of gene expression on silk proteins and protein structure and, ultimately, silk functional properties to understand the mechanisms inducing nutritionally induced property variation in spider silks at multiple scales.

Here we performed an examination of nutritionally induced MA silk property variation by running experiments similar to those described by Blamires et al. [27]. However, we used five species of Araneoid spiders: Argiope keyserlingi, Eriophora transmarina, Latrodectus hasselti, Nephila plumipes and Phonognatha graeffei, and directly measured spidroin expression using...
quantitative real-time PCR (RT-PCR) techniques. Specifically, our experiments aimed to answer two persistent and problematic questions about spider silk property variability: (1) Do the silk nanostructures and mechanical properties of different spiders respond similarly to variations in spidroin expression? And (2) what are the relative contributions of changes to amino acid compositions and nanostructures in inducing spider silk mechanical property variation? To answer the first question, we performed silk tensile tests, SAXS/WAXS analyses, amino acid determinations, and gene expression analysis, for the abovementioned spiders and compared the results across species and treatments. To answer the second question, we pooled the mechanical property, nanostructural, and amino acid compositional data across species and constructed predictive models.

Materials and methods

Ethics statement

Ethical clearance was not required to perform this research. Capture permits were not required under New South Wales law as all collections were made outside of protected areas. We confirm that the collection locations were not privately owned, and we did not collect any endangered or protected species.

General methods. Fig 1 overviews the methods implemented to, collectively, examine the consequences of gene expression on silk protein nanostructures and functional properties.

Spider collection and pre-feeding. We collected, per species, 40 adult female *A. keyserlingi*, *E. transmarina*, *L. hasselti*, *N. plumipes* and *P. graeffei* from locations between Sydney and Ballina, New South Wales, Australia, during trips made between October 2014 and January 2015. To ensure spiders of approximately equal size within species were used and that no gravid spiders were collected, we measured each spider’s body length and width to ±0.1 mm using digital Vernier calipers (Caliper Technologies Corp., Mountain View, CA, USA) and mass to ±0.001 g using an electronic balance (Ohaus Corp., Pine Brook, NY, USA) upon collection, and discarded any particularly large or heavy (>50% above the mean) individuals. We returned all the required spiders to the laboratory at the University of New South Wales, Sydney, where they were placed in 115 mm (wide) x 45 mm (high) plastic circular containers. The containers had perforated wire mesh lids with a 20 mm long slit cut into them using a Stanley

---

Fig 1. Flow diagram of the methods implemented to examine the consequences of protein deprivation on MA silk gene expression, protein nanostructures, and mechanical properties for five species of spiders.

https://doi.org/10.1371/journal.pone.0192005.g001
knife to facilitate feeding with a 50 μl micropipette. We pre-fed the spiders 20 μl (A. keyserlingi, L. hasselti and P. graeffei) or 50 μl (E. transmarina and N. plumipes) of a 30% (w/v) glucose solution daily over five days (for details see [27,51]) to standardize the diet of all spiders prior to experimentation. We reweighed the spiders after the pre-feeding treatment and any individuals who lost > 50% of their initial mass (one A. keyserlingi, one L. hasselti and three N. plumipes) were discarded.

Feeding experiments
We randomly divided the remaining 40 or so spiders per species equally into two groups and fed them either one of two solutions over 10 days: a protein solution (P) or protein deprived solution (N). The protein solution used to experimentally increase protein intake was identical to that used by Blamires et al. [27], i.e. a mixture of 10g of a 10% albumin solution with 6g of sucrose in 60ml of water. The protein deprived solution was 8g of sucrose in 30 ml of water. We fed the spiders by placing a measured droplet of solution onto their chelicerae using a 20 μl micropipette (see [27,51]). As protein and carbohydrates contain approximately similar energy densities (~4kJ g⁻¹), solutions of similar energy concentrations were fed to all spiders. After completing the feeding experiment we re-weighed all of the spiders and any that lost > 20% of their mass during the experiment (i.e. one A. keyserlingi, two L. hasselti and P. graeffei, and five N. plumipes) were not used for any of the subsequent experiments.

Silk collection
After completing the feeding experiment and subsequent re-weighing, we anaesthetized each spider (N = 185 spiders; 38 A. keyserlingi, 40 E. transmarina, 37 L. hasselti, 32 N. plumipes, and 38 P. graeffei) using CO₂ and carefully pulled a single MA silk fiber from their spinnerets using tweezers. We collected a thread of silk from eight spiders per treatment per species for the determination of mechanical properties as follows.

We connected a revolving headframe to an electronic spool. We then attached a 240 mm long x 40 mm wide cardboard strip with six 10 mm x 10 mm square holes punched at 30 mm intervals to the headframe. Double sided sticky tape was stuck onto the cardboard at the border of the holes. A silk thread was pulled over the headframe and stuck to the sticky tape. The headframe was rotated once at 1m min⁻¹ while ensuring the silk traversed all of the square holes and adhered to the pieces of tape. The strip was then removed from the headframe and a drop of water based glue applied at the position where the silk attached to the tape. Another frame of equal size with identically positioned holes punched into it was placed on top. The two strips were squeezed together with forceps ensuring that they were tightly stuck together. We then cut the strip in the regions between the holes perpendicular to the silk thread, thus leaving six 10mm x 10mm frames each holding parts of a single thread of silk. The procedure was repeated for every individual used from each of the five species. Accordingly, 48 frames were collected per treatment per species, i.e. 6 frames x 8 individual threads (see [26,52] for details).

We collected silk from a further three individuals per treatment per species for SAXS/WAXS analyses. We did this by spooling their silk onto 3 mm x 1 mm titanium frames containing 0.5 mm x 0.5 mm windows (see [52]) for ~1 h (A. keyserlingi, L. hasselti and P. graeffei) or ~2 h (E. transmarina and N. plumipes). We assumed that the amount of silk extracted was approximately the entire store of silk from the spider’s major ampullate glands. We collected between 1000–2000 rounds of silk across the windows of each frame. We have previously found this amount of silk to be adequate for attaining quality scattering from 5-35keV synchrotron X-ray sources [25,27,52].
We collected silk from the remaining 75 spiders [16 *A. keyerslingi* (8 each from the N and P treatment), 18 *E. transmarina* (9 N and 9 P), 15 *L. hasselti* (7 N and 8 P), 10 *N. plumipes* (4 N and 6 P) and 16 *P. graeffei* (8 N and 8 P)] to determine their amino acid compositions. We wrapped the silk threads around a glass tube connected to the electronic spool spun at 1 m min\(^{-1}\) for ~1–2 h. This approximated the collection of the store of silk from the spider’s glands, so any variations in amino acid composition within individual threads were accounted for. All silks were extracted under controlled temperature (~25˚C) and humidity (~50% R.H.) in still air, so reeling speed and post-spin environment did not influence the subsequent chemical or mechanical property measurements.

**Mechanical property determination**

One randomly selected frame of silk from each thread (i.e. one of the six frames of silk collected per spider) was used to ascertain the width of the thread \([26,52]\) so we could calculate the cross-sectional area of the individual threads used in the ensuing tensile tests. We taped the frames to a microscope slide and examined and photographed them under air immersion at 1000x magnification using a polarized light microscope (CKX41, Olympus, Tokyo, Japan) connected to a SPOT Idea 5 Mp digital camera (Spot Imaging Solutions, Sterling Heights, MI, USA). The images were digitized using the program Spot Basic 4.7 (Spot Imaging Solutions, Sterling Heights, MI, USA) and the width of each thread determined as a mean of 12 measurements using the program Image J (NIH, Bethesda MD, USA).

We performed the following tensile tests under controlled temperature (~25˚C) and humidity (~50% R.H.) in still air within 10 days of silk collection. We placed each of the cardboard frame-mounted silks for each species within the grips of an Instron 5543 tensile testing machine (Instron Machines, Melbourne, Australia) with a ~2 μN resolution \([26]\). We ensured that the grips held the silks firmly at the upper and lower frame edges. The left and right sides of the frames were cut away and the silks stretched at a rate of 0.1 mm s\(^{-1}\) until the fiber ruptured.

True stress (\(\sigma\)) and strain (\(\epsilon\)) were derived from the following equations:

\[
\sigma = \frac{F}{A}
\]

and

\[
\epsilon = \log \frac{L}{L_0}
\]

where F is the force applied to the specimen, A is the cross-sectional area of the thread calculated from the thread diameter assuming a constant thread volume, L is the instantaneous length of the fiber at a given extension value and \(L_0\) is the original gage length of the fiber. Stress vs strain curves were determined for each silk tested by a standard trapezoidal method from which we calculated the following mechanical properties using both Bluehill 3.0 (Instron Machines, Melbourne, Australia) and Microsoft Excel 2010: (1) ultimate strength; or the stress at rupture, (2) extensibility; or the strain at rupture, (3) toughness; the area under the stress strain curve, and (4) Young’s modulus (stiffness); the slope of the stress-strain curve during its initial elastic phase.

**SAXS/WAXS experiments**

Small-angle X-ray scattering (SAXS) procedures were performed at the end station of beamline BL23A SWAXS of the Taiwan Light Source, National Synchrotron Radiation Research Center.
Pre-tests were performed using polyethylene and silver behenate to calibrate the scattering intensity and wave vector sensitivity, respectively. The samples were placed 3659 mm from the incident 15 keV X-ray beam ($\lambda = 0.8265$ Å) and were exposed to the beam for 10–60 s depending on the measured signal intensity. The scattered radiation was captured using a Pilatus 1M-F area detector and two dimensional SAXS images generated. From these images the scattering intensity was obtained and intensity vs scattering vector ($q$) plots generated using the program Albula (Dectris, Baden-Dättwil, Switzerland). Where $q$ was ascertained by the equation:

$$q = 4\pi \lambda^{-1} \sin \theta$$

where $\lambda$ is the wavelength of the incident X-ray beam ($\lambda = 1.033$ Å) and $\theta$ is the scattering angle [53].

We then calculated the meridional peak following Balta-Calleja and Vonk [54], from which we estimated the long period ($L$) using the equation [54–56]:

$$L = 2\pi / q_m$$

where the $q_m$ is the integrated position of the meridional peak ascertained by scanning the intensity vs $q$ plots along the equatorial direction [34].

We performed WAXS procedures immediately upon completion of the SAXS procedures using the same 30 silk samples at the end station of beamline BL01C2 at NSRRC, Hsinchu, Taiwan. We used the same samples because we were interested in measuring a combination of nanostructural properties within the crystalline, amorphous and lamella regions of the same silk threads. We thus first performed a series of pre-tests to establish that the short exposure time (10–60 s) of the SAXS procedures was unlikely to damage the silk nanostructures and affect the WAXS measurements.

We placed the samples 300 mm from the 12keV incident beam and exposed them to X-rays for 10–60 min depending on the pre-measured signal intensity. Beam size was confined by a collimator 0.5 mm in diameter. Scattered radiation was detected by a Mar 345 imaging plate and two dimensional diffraction images generated for each silk sample using the program Fit2D (ESRF, Grenoble, France). From the diffraction images we calculated the diffraction angle ($\theta$), intensity peaks ($I_x$), and 2$\theta$ full width and half width maximum intensities (FWHM) of the (200) and (120) diffraction peaks and the so called amorphous halo.

We subsequently calculated:

(i) The relative crystalline intensity ratios ($I_{200}/I_{120}$) with $I_{200}$ and $I_{120}$ representing the sum of the intensity peaks at the (200) and (120) peaks respectively [57].

(ii) The crystallinity index, $X_c$, calculated according to Grubb and Jelinski [58], and

(iii) Herman’s orientation function, $f_c$, using the equation [42]:

$$f_c = (3\{\cos^2\theta\}1)/2$$

where $\varphi$ is the angle between the c axis and the fiber axis, $\{\cos^2\varphi\}$ is the azimuthal width at the (200) and (120) diffraction peaks determined using the equation [42]:

$$\{\cos 2\varphi\} = 1 - A\{\cos 2\varphi_1\}B\{\cos 2\varphi_2\}$$

where $A = 0.8$ and $B = 1.2$. 
Amino acid composition determination

We weighed all of the silk samples designated for amino acid composition analysis to the nearest 0.001 mg on an electronic balance (Pioneer PA214C, Ohaus, Pine Brook NJ, USA), before submergence in 99% hexofluoro-isopropanol solvent (500 μl of per mg of silk) within 1 ml Eppendorf tubes. The samples were then hydrolyzed in 6 mol l⁻¹ HCl for 24 h at 115˚C. Molar percentage compositions of glutamine, serine, proline, glycine, and alanine, the amino acids representing ~90% of the total amino acids in the MA silks of most spiders [59], determined using an Alliance Systems (Waters, Rydalmere NSW, Australia) high performance liquid chromatography column [60] at the Australian Proteomic Analysis Facility, Sydney.

Spidroin gene expression analysis

At the end of silk collecting five randomly selected spiders per treatment and species were sacrificed by lethal exposure to CO₂ and their major ampullate glands dissected as described by Jeffrey et al. [61]. The glands and a sample of the remaining abdomen were immediately lysed with RNase free mini pestles in QIAzol Lysis Reagent and mRNA extracted using an RNeasy Plus Universal RNA extraction kit (Qiagen, Düsseldorf, Germany). In order to prevent any DNA contamination we used a gDNA Eliminator Solution provided with the extraction kit to remove all genomic DNA. The extracted mRNA was eluted to 30–35 μl and we measured the molar concentration using a NanoDrop 1000 Spectrophotometer (Thermo Fisher Scientific, Wilmington, DE, USA). A mean concentration of 1376.4 ± 239.04 ng/μl RNA was extracted from all samples and did not differ substantially between species. Mean absorbance ratios of the samples were 2.24 at 260/280 nm, and ranged between 1.86 and 2.38, and 1.97 at 260/230 nm and ranged between 1.22 and 2.27. Absorbance ratios in this range are considered acceptable as ‘pure’ for single stranded RNA [62].

All of the mRNA samples were diluted to 1000 ng/μl. We took 12.5 μl subsamples of the diluted mRNA for reverse transcription to cDNA using an Advantage reverse transcription kit for PCR (Clontech, Clayton Vic, Australia). The reverse transcription and PCR activation procedures were carried using Eppindorf Mastercycler (Eppindorf, Mamburg, Germany) qPCR machines, following the recipe outlined by the reverse transcription kit handbook [63]. A “buffers only” (i.e. no RNA and no Reverse Transcriptase) solution was included in the analyses as a negative control. We included the spider’s abdomens in the gene expression analysis for normalization against background expression of MaSp transcripts in other silk glands or abdominal tissue [10]. All procedures were replicated three times for each individual spider.

We used the *Drosophila rufa* Glycerol-3-phosphate dehydrogenase (g3pdh) gene as a “housekeeping” reference gene for our RT-PCR analyses, as is common practice in gene expression analyses [64,65]. However, since the g3pdh gene primers were designed from *Drosophila* spp., there may have been unintended amplification biases [66] across species.

We diluted all of the cDNA eluent to 200 ng/μl, checking (using a NanoDrop 1000 Spectrophotometer) that the 260:280 nm absorbance ratio were all between 1.5 and 1.8 before sending 10 μl samples to the Ramaciotti Centre for Genomics, University of New South Wales, for Fluidigm quantitative RT-PCR gene expression analyses [67]. Fluidigm RT-PCR gene expression analysis utilizes dual-labelled probes designed to hybridize a complementary region of the cDNA for real-time amplification. These probes contain a fluorescent reporter dye on the 5’ base, and a quencher on the 3’ base, whose intensity increases proportional to the number of probe molecules cleaved [67].

We used published C-terminal domain sequences for MaSp1 and MaSp2 from *Argiope trifasciata* (hereon called MaSp1a and MaSp2a) and *Latrodectus hesperus* (hereon called MaSp1b
...and MaSp2b) [768–71] to order primers for the RT-PCR analyses (see Supporting Information, S1 Table for sequences and accession numbers) using the Fluidigm online assay designer [67]. We converted the threshold cycle (C_T) values (see S2 Table) derived by the RT-PCR analyses to 2^{-ΔΔC_T} values, which were averaged for each individual spider across the three technical replicates, following Schmittgen and Livak [62].

While recent genomic work has shown that some spider species may possess more than two MaSp loci, with some paralogs exhibiting different expression patterns across different silk glands [10,68,72], further verification of multiple MaSp loci across the Araneoid clade awaits. Since our objective here was to identify whether shifts in amino acid compositions can be attributable to shifts in MaSp1: MaSp2 expression and not verify or refute the multiple loci hypothesis, we used the abovementioned procedures to ascertain the across treatments expression patterns of just two MaSp1 and MaSp2 paralogs.

Analyses

For each species we used separate single-factor (two treatment levels: protein deprived and protein fed) multivariate analyses of variance (MANOVAs) and Fisher’s Least Significant Difference post-hoc analyses to determine whether the mean (± 1 standard error): (1) mechanical properties (ultimate strength, extensibility, toughness, and Young’s modulus), (2) nanostructures (L, 2θ FWHM of the (200) and (120) diffraction peaks and amorphous halo, I_{200}/I_{120}, X_c, and f_c), (3) mole compositions of glutamine, serine, proline, glycine, and alanine, and (4) spidroin (MaSp1a, MaSp1b, MaSp2a, and MaSp2b) expression. We used additional univariate (treatment) ANOVAs to individually compare the MaSp1a, MaSp1b, MaSp2a, and MaSp2b 2^{-ΔΔC_T} values between treatments for each species. We log_{10} or arcsine (amino acid composition data) transformed any data that failed Levene’s heterogeneity tests.

To ascertain the influences of nanostructures and amino acid compositional variations on silk mechanical properties we pooled the data for all species and constructed multiple regression models. We used the mechanical properties that our MANOVAs found to differ across treatments as the response variables and any nanostructural parameters or amino acid compositions that our MANOVAs found to differ across treatments as the predictor variables. Species and treatments were assigned as continuous predictor variables. A large number of predictor variables, interactions, and intercept terms were likely so we considered linear regression models to be too complex for interpretation. We therefore derived Y = β_0 + β_1(x_1) + β_2(x_2) + ... + β_n(x_n) + e_i additive regression models for each response variable, where β_0 is the population intercept, β_1, β_2, ... , β_n are the regression coefficients associated with the predictor variables x_1, x_2, ... , x_n, and e_i is the random error term associated with the i_th observation [73]. We checked all data for normality, linearity, homoscedasticity and singularity using Q–Q and scatterplots prior to constructing the models.

Results

An overall summary of the results for the five species of spider is provided in Table 1.

Does the silk nanostructures and mechanical properties of different spiders respond similarly to variations in spidroin expression?

We found that protein feeding and deprivation affected silk mechanics differently among the five species examined (see Supporting Information, S3 Table). Argiope keyserlingi’s silk was more extensible when protein deprived than when protein fed. On the other hand Eriophora transmarina’s silk was less extensible when protein fed (Supplementary Material, S3A and S3B
Both *Nephila plumipes*’ and *Phonognatha graeffei*’s MA silks were stronger and tougher when they were protein deprived (Supporting Information, S3D and S3E Table). Wherein we found neither protein feeding nor deprivation to effect the mechanical properties of *Latrodectus hasselti*’s silk (Supporting Information, S3C Table). Comparisons of the silk thread widths across treatments found a significant difference for *A. keyserlingi* only (means ± SE: protein deprived spiders = 3.39 ± 0.18 μm, protein fed spiders = 2.17 ± 0.15 μm, Supporting Information, S3A Table), thus thread width differences across treatments were not responsible for any of the variations in silk mechanical properties found across the five species.

The silk nanostructures varied in response to protein feeding/deprivation among the five spiders (see Supporting Information, S4 Table). The SAXS images and subsequent intensity vs scattering vector (q) plots for each spider are shown in Fig 2 and Fig 3 respectively. Examples of two dimensional WAXS images are shown in Fig 4 and the subsequent intensity vs 2θ plots in Fig 5. The azimuthal angles at the (200) and (120) diffraction peaks are in the Supporting Information (see S1 Fig and S2 Fig, respectively). From the various plots we calculated a greater long period in *A. keyserlingi* silks when the spiders were protein deprived compared to when they were protein fed. Within species, the nanostructures generally shifted in the same direction as the mechanical properties we predicted them to affect, e.g. long period and/or FWHM of the amorphous halo varied with extensibility in *A. keyserlingi* and *N. plumipes* (Supporting Information, S4 Table). We thus expect the structural variations to explain much of the variation in mechanical properties.

The FWHM of the (200) peak was greater in *A. keyserlingi* silks when protein deprived compared to when protein fed. Crystallinity, on the other hand, was greater in the silks of protein fed *A. keyserlingi* (Supporting Information, S4A Table). These findings and the greater long period in the silk of protein deprived *A. keyserlingi* indicate that the crystalline nanostructures in their silks were stacked more densely when the spiders were protein fed, and were more aligned and stretched when they were protein deprived. We considered it likely that the greater alignment of the crystalline region proteins in the silks of protein deprived *A. keyserlingi* explains the high extensibility of their silks. FWHM at the amorphous halo was greater in the silks of protein deprived compared to protein fed *N. plumipes* (S4D Table). This result, and the greater ultimate strength in the silk of protein fed *N. plumipes*, indicated that variability in the non-crystalline nanostructures primarily influenced their silk mechanical properties. SAXS/WAXS did not detect any significant nanostructural shifts across treatment for *E. transmarina*, *L. hasselti* and *P. graeffei* silks (Supporting Information, S4B, S4C and S4E Table).

Amino acid compositions of the MA silks varied across treatments and the type of variations found differed for each of the spiders. For instance, we found a reduction in the proline, alanine and glycine compositions of *A. keyserlingi* silk when protein deprived (Supporting Information, S4D Table).

### Table 1. Summary of the consequences of protein deprivation on silk properties for the five species examined.

|                        | Argoipe keyserlingi | Eriophora transmarina | Latrocedtus hasselti | Nephila plumipes | Phonognatha graeffei |
|------------------------|---------------------|-----------------------|----------------------|------------------|----------------------|
| **Mechanical properties** |                     |                       |                      |                  |                      |
| thread width           | —                   |                       |                      | —                |                      |
| extensibility          | —                   |                       |                      | —                |                      |
|                        | —                   |                       |                      | —                |                      |
| **Crystalline structures** |                     |                       |                      |                  |                      |
| crystallinity          | —                   |                       |                      | —                |                      |
|                        | —                   |                       |                      | —                |                      |
| **Non-crystalline structures** |                   |                       |                      |                  |                      |
| lamellar alignment     | —                   |                       |                      | —                |                      |
|                        | —                   |                       |                      | —                |                      |
| **Amino acid composition** |                     |                       |                      |                  |                      |
| Gly, Ala               | —                   |                       |                      | —                |                      |
|                        | —                   |                       |                      | —                |                      |
| **MaSp expression**    | Upregulate MaSp1a   | Downregulate MaSp1a   | Upregulate MaSp1a, MaSp2a, MaSp2b | Downregulate MaSp1a, MaSp2a | Downregulate MaSp1a, Upregulate MaSp2a |

https://doi.org/10.1371/journal.pone.0192005.t001

*Table*. Both *Nephila plumipes*’ and *Phonognatha graeffei*’s MA silks were stronger and tougher when they were protein deprived (Supporting Information, S3D and S3E Table). Wherein we found neither protein feeding nor deprivation to effect the mechanical properties of *Latrodectus hasselti*’s silk (Supporting Information, S3C Table). Comparisons of the silk thread widths across treatments found a significant difference for *A. keyserlingi* only (means ± SE: protein deprived spiders = 3.39 ± 0.18 μm, protein fed spiders = 2.17 ± 0.15 μm, Supporting Information, S3A Table), thus thread width differences across treatments were not responsible for any of the variations in silk mechanical properties found across the five species.
A reduction in proline composition was, however, detected in *N. plumipes* and *P. graeffei* MA silks when they were protein fed. Fig 6 shows mean spidroin expression ($2^{-\Delta \Delta CT}$) values across treatments for each species. Interestingly, the across treatment expression patterns differed significantly among the five species (Wilk's $\lambda = 0.011$, d.f. = 5,10, $P < 0.001$). We found that *A. keyserlingi* upregulated the expression of *MaSp1a* within their MA gland when protein deprived ($F_{1,28} = 13.911$, $p < 0.001$). *E. transmarina* on the other hand downregulated the *MaSp1a* expression within their MA gland when protein deprived ($F_{1,28} = 42.171$, $p < 0.001$), but upregulated *MaSp1b* ($F_{1,28} = 8.135$, $p = 0.005$). *L. hasselti* upregulated their *MaSp1a* ($F_{1,28} = 12.308$, $p < 0.001$), *MaSp2a* ($F_{1,28} = 27.604$, $p < 0.001$), and *MaSp2b* expression ($F_{1,28} = 25.342$, $p < 0.001$) expressions when protein deprived, while downregulating *MaSp1b* ($F_{1,28} = 54.224$, $p < 0.001$), while *N. plumipes* (c.f. response to protein deprivation by *N. pilipes* [27]) downregulated both their *MaSp1a* ($F_{1,28} = 21.104$, $p < 0.001$) and *MaSp2a* ($F_{1,28} = 8.358$, $p = 0.007$) expression when protein deprived, which corresponded with an increase in proline composition. *P. graeffei* significantly downregulated their *MaSp1a* ($F_{1,28} = 6.587$, $p = 0.010$) and upregulated their *MaSp2a* ($F_{1,28} = 8.543$, $p = 0.006$) expression when protein deprived.

**What are the contributions of changes in amino acid compositions and nanostructures in inducing spider silk mechanical property variation?**

To answer this question we derived additive regression models for three response (Y) variables pertaining to the mechanical properties: ultimate strength, extensibility and toughness, and
Fig 4. WAXS images derived for MA silks of each species of spider. Where P = protein fed and N = protein deprived treatments, Ak = Argiope keyserlingi, Et = Erisophora transmarina, Lh = Latrodectus hasselti Np = Nephila plumipes, Pg = Phonognatha gracilis.

https://doi.org/10.1371/journal.pone.0192005.g004
eight predictor ($x_i$) variables: thread width ($x_1$), FWHM of the amorphous halo ($x_2$), FWHM of the (200) diffraction peak ($x_3$), crystallinity ($x_4$), long period ($x_5$), proline composition ($x_6$), glycine composition ($x_7$), and alanine composition ($x_8$) using data from all species excluding $L$. hasselti. The variables chosen were those that our analyses above found to differ across treatments in at least one species. Our models showed crystallinity to predominantly influence the ultimate strength in nutritionally affected MA silks (Table 2A). Long period, which represents the nanostructure alignment across the silk’s amorphous and lamellar regions, and alanine and proline composition, were predominantly influential over extensibility (Table 2B). Long period, along with glycine, alanine and proline compositions, influenced silk toughness (Table 2C).

**Discussion**

Our study was the first to comprehensively examine the consequences of protein deprivation on variations in spider silk mechanics, structures, amino acid compositions and gene expressions at multiple scales across species. We concluded that: (i) MA silk properties of the spiders respond differently across multiple scales to variations in nutritional intake, and (ii) variations in spidroin expression and the crystalline and non-crystalline nanostructures play specific roles in inducing variations in the silk’s mechanical properties.

The *MaSp1: MaSp2* expression patterns we found across treatments generally did not correspond with the amino acid compositions according to our expectations under a traditional *MaSp* expression model. For instance, a decreased in alanine and glycine composition in the MA silk of protein deprived *A. keyserlingi* was not associated with any form of *MaSp2* downregulation, but was associated with an upregulation of *MaSp1a*. The spidroin expression pattern found for *E. transmarina* across treatments likely explains why the amino acid compositions did not vary across treatments in this spider. The high expression of all of the *MaSp1* and *MaSp2* paralogs in the MA glands of *L. hasselti* is consistent with findings for *L. hasperus* [24,68,74]. Expression of multiple *MaSp* paralogs in the MA glands was not common for most species we assessed, but was pronounced in *L. hasselti*. The increase in proline composition in protein deprived *N. plumipes* MA silk conflicts with the findings of Blamires et al. [27], who
found an increase in proline composition in protein deprive *N. pilipes*. We would have expected the downregulation of *MaSp2a* expression by this spider when protein deprived to result in a decrease in silk proline composition. The downregulation of *MaSp1a* on the other hand was unlikely to have any influence on proline composition [51,75,76]. An increase in the proline composition in the silks of protein deprived *P. graeffei*, however, may be ascribed to an increased in *MaSp2a* expression by these spiders.

The apparent disconnect between the spidroin expression patterns and the amino acid compositions in the silks of most species might lead us to conclude that: (1) the silk nanostructures and mechanical properties of different spiders do not respond similarly to variations in spidroin expression, and (2) the relative contributions of change in spidroin expression in inducing spider silk mechanical property variation is minimal at best. We, nevertheless, presume caution before drawing definitive conclusions about the influence of *MaSp* expression in light of the following. Firstly, in our RT-PCR reactions any specific amplification of *MaSp* paralogs could arise in any given species using a single pair of primers [65]. This may cause falsely elevate expression levels of certain *MaSp* transcripts. Examination of the melt curves for the genes screened suggested that in some species there were indeed multiple amplifications.

Fig 6. Comparisons of the expressions of the *MaSp1* genes previously isolated from the *Argiope trifasciata* (*MaSp1a*) and *Latrodectus hesperus* (*MaSp1b*), and the *MaSp2* genes from *Argiope trifasciata* (*MaSp2a*) and *Latrodectus hesperus* (*MaSp2b*), across treatments for each of the five spiders. Where P = protein fed and N = protein deprived treatments, Ak = Argiope keyserlingi, Et = Eriophora transmarina, Lh = Latrodectus hasselti Np = Nephila plumipes, Pg = Phonognatha graeffei.

https://doi.org/10.1371/journal.pone.0192005.g006
The most discernible example of this is found in the melt curves for MaSp1a and MaSp2a, particularly for Nephila plumipes (see S3 Fig). Secondly, the amplification of non-orthologous MaSp loci across species with single primer pairs could lead to falsely elevated or falsely lowered detection of expression levels [77]. Any of the MaSp-targeting primers could thus have identified a single MaSp transcript in one species, a single non-orthologous transcript in a second species, and/or multiple non-orthologous transcripts in yet another species. Thirdly, since the full length MaSp sequences for each of the species under investigation herein is not known, we designed primers based on the MaSp sequences for closely related species (Argiope trifasciata and Latrodectus hesperus). The amplification efficiency thus may potentially be biased toward species with the greatest sequence homology (i.e. Argiope keyserlingi and Latrodectus hasselti). Indeed only in L. hasselti was expression of MaSp1b and MaSp2b comparable to that of MaSp1a and MaSp2a. Lastly, some degree of gDNA amplification cannot be ruled out as

| (a) Ultimate strength | Predictor variable | b estimate | Standard error of b | Standard co-efficient | t₁ | P-value |
|----------------------|-------------------|------------|---------------------|----------------------|----|---------|
| Intercept            | 10829             | 3517.103   | 0                   | 3.070                | 0.054 |
| thread width (x₁)    | 2.448             | 1.877      | 0.030               | 1.305                | 0.283 |
| FWHM: amorphous halo (x₂), | 89.601 | 0.971 | -0.375 | -2.622 | 0.078 |
| FWHM: (200) (x₃)     | -368.230          | 1.874      | -0.272              | -1.214               | 0.311 |
| Xₜ (x₄)              | 2.761             | 1.001      | 0.158               | 3.756                | 0.040 |
| L (x₅)               | -14.814           | 4.081      | -0.056              | -3.031               | 0.056 |
| % proline (x₆)       | 57.631            | 35.095     | 0.631               | 1.639                | 0.348 |
| % glycine (x₇)       | 0.880             | 29.037     | 0.023               | 0.230                | 0.081 |
| % alanine (x₈)       | 53.030            | 24.886     | 0.289               | 2.157                | 0.176 |

| (b) Extensibility    | Predictor variable | b estimate | Standard error of b | Standard co-efficient | t₁ | P-value |
|----------------------|-------------------|------------|---------------------|----------------------|----|---------|
| Intercept            | 0.656             | 0.388      | 0                   | 1.771                | 0.327 |
| thread width (x₁)    | -0.061            | 0.027      | -0.696              | -2.273               | 0.163 |
| FWHM: amorphous halo (x₂), | 0.159 | 0.058 | 0.395 | 2.720 | 0.104 |
| FWHM: (200) (x₃)     | -0.188            | 0.038      | -0.031              | -1.214               | 0.429 |
| Xₜ (x₄)              | 0.154             | 0.019      | 0.132               | 1.167                | 0.342 |
| L (x₅)               | 0.018             | 0.005      | 0.315               | 3.784                | 0.020 |
| % proline (x₆)       | 0.071             | 0.014      | 0.472               | 4.089                | 0.012 |
| % glycine (x₇)       | 0.026             | 0.011      | 0.360               | 2.218                | 0.065 |
| % alanine (x₈)       | -0.044            | 0.010      | 0.502               | -3.354               | 0.043 |

| (c) Toughness        | Predictor variable | b estimate | Standard error of b | Standard co-efficient | t₁ | P-value |
|----------------------|-------------------|------------|---------------------|----------------------|----|---------|
| Intercept            | -80.046           | 44.049     | 0                   | -1.817               | 0.120 |
| thread width (x₁)    | -27.372           | 3.041      | 0.135               | -2.447               | 0.067 |
| FWHM: amorphous halo (x₂), | 68.411 | 6.639 | 0.168 | 2.304 | 0.061 |
| FWHM: (200) (x₃)     | -12.316           | 3.499      | 0.080               | -1.519               | 0.176 |
| Xₜ (x₄)              | 25.414            | 10.463     | 0.056               | 2.248                | 0.084 |
| L (x₅)               | 4.259             | 0.671      | 0.134               | 5.351                | 0.009 |
| % proline (x₆)       | 23.488            | 1.640      | 0.201               | 7.312                | <0.001 |
| % glycine (x₇)       | 14.651            | 1.630      | 0.153               | 5.853                | 0.005 |
| % alanine (x₈)       | -12.591           | 1.517      | 0.124               | -3.924               | 0.015 |

* denotes the significantly influential predictor variable(s) for each model.

https://doi.org/10.1371/journal.pone.0192005.1002

The most discernible example of this is found in the melt curves for MaSp1a and MaSp2a, particularly for Nephila plumipes (see S3 Fig). Secondly, the amplification of non-orthologous MaSp loci across species with single primer pairs could lead to falsely elevated or falsely lowered detection of expression levels [77]. Any of the MaSp-targeting primers could thus have identified a single MaSp transcript in one species, a single non-orthologous transcript in a second species, and/or multiple non-orthologous transcripts in yet another species. Thirdly, since the full length MaSp sequences for each of the species under investigation herein is not known, we designed primers based on the MaSp sequences for closely related species (Argiope trifasciata and Latrodectus hesperus). The amplification efficiency thus may potentially be biased toward species with the greatest sequence homology (i.e. Argiope keyserlingi and Latrodectus hasselti). Indeed only in L. hasselti was expression of MaSp1b and MaSp2b comparable to that of MaSp1a and MaSp2a. Lastly, some degree of gDNA amplification cannot be ruled out as
influencing our results. We, however, did not expect this to cause major expression value biases within any particular species.

While the above caveats may tempt us to think that the RT-PCR analyses yielded largely uncertain results, we found that the expression patterns for *P. graeffei* aligned exceptionally close to our expectation should the spiders be regulating *MaSp1* and *MaSp2* expression alone. We expect that such a finding would not have been possible if our primer choices and/or amplifications were compromised in any way. Moreover, our expression values conformed with those reported for the *MaSp*-a, *MaSp*-f, and *MaSp*-g genes by Babb et al. [10] (although these authors reported a wider range of values for other *MaSp* loci). We thus expect that the across-treatment expression patterns that we reported for each species to be reliable for the explicit purpose of checking *MaSp* expression against the amino acid compositions for each species.

Our study corroborates work showing that amino acid compositional shifts confer consequences on crystalline and non-crystalline protein structures [11,17,26,27,49] and alignments within the crystalline, amorphous and lamellar regions of the silk, which in turn influences its strength, extensibility and toughness [1,40,43,50,52]. Our study also uncovered novel mechanisms behind the multilevel shifts in silk properties, as follows.

While silk mechanical properties were affected by protein deprivation in four of the five species of spider (the exception being *L. hasselti*), the mechanisms by which the mechanics were affected differed in each instance. Our amino acid composition analyses, for instance, found a lowering of proline, alanine and glycine compositions in the silk of *A. keyserlingi* when protein deprived. Nevertheless the silks of *N. plumipes* and *P. graeffei* had greater proline compositions when protein deprived. Our genetic expression analyses revealed that most spiders seemed to preferentially regulate their expression of the *MaSp1* genes rather than *MaSp2* genes across treatments, contrary to what might be predicted, given the *MaSp2* protein is expected to be the more costly of the two proteins to metabolically synthesize [27,51,75].

Our SAXS measurements revealed that the spidroin regulation and consequent variability in silk proline corresponded with variations in long period (*L*) in *A. keyserlingi*, *N. plumipes* and *P. graeffei*, suggesting that spidroin expression and/or proline composition affects nanostructural alignment in the amorphous and lamellar regions. These compositional and structural variations correlated well with extensibility and toughness variations in the silks, providing first evidence of a functional link between spidroin expression, nanostructural formation, and mechanical property variations in the MA silks of different spiders.

An explanation for the association between proline and amorphous region nanostructural alignment might lie in the capacity for proline to form cross-linkages which disrupt the hydrogen bonds between amorphous region α-helices and other structures causing slippage in the nanostructures under strain [20]. Our finding of a lack of any change in amorphous and lamella region alignment in the silks of protein deprived *P. graeffei*, despite significant variations in proline composition, across treatments nonetheless suggests that proline does not necessarily directly affect amorphous region nanostructural alignment but likely provides the conditions for the breaking and re-establishment of hydrogen bonds in the region [78].

Our amino acid composition and WAXS analyses, and subsequent modelling, predicted that, in contrast to extensibility and toughness, MA silk ultimate strength was primarily influenced by variations in crystallinity independent of *MaSp1*: *MaSp2* expression or any subsequent shifts in amino acid composition. Our WAXS analyses withal suggested that variations in amorphous region nanostructural alignment and crystallinity combine to influence silk strength. Glandular pH, salts and shear stresses during spinning influence the formation of the crystalline nanostructures [11]. Hence, we deduced that between treatment variations in glandular pH, salts and shear stress induce the crystalline region proteins to undergo α-helix→β-
sheet nanostructural phase transitions within the spinning duct [79]. These transitions result in an increase in crystalline density [50], which causes the crystalline region nanostructures to realign under strain [1]. Such an increase in crystalline density should be identifiable in WAXS experiments as a reduction in $2\theta$ and FWHM at the (200) or (120) diffraction peaks [32]. Indeed, our analyses herein revealed such a mechanism occurred in protein deprived $N.\ plumipes$ silks (S1 Table). Nonetheless, the phenomenon was not detected in the silks of any other species, leading us to conclude that the precise physiological mechanisms inducing property variation differed among the five species.

Unlike the other four species, we found that neither protein feeding nor deprivation effected the amino acid compositions, nanostructures, or mechanical properties of $L.\ hasselti$'s MA silk. A similar lack of variability in MA silk structure and mechanical properties was found in $L.\ hasselti$ collected at different times of year [26]. Spiders of the genus $Latrodectus$ use MA silk within their three-dimensional cobwebs as structural supports [80], whereas the orb web building spiders use MA silks within webs to absorb the impact of flying prey [11]. A testable hypothesis might accordingly be that orb web spider MA silks have a greater inherent variability in order to adjust the functionality of the orb web [81] and this variability is triggered by changes in nutrient uptake. The MA silks of cobweb spiders on the other hand do not require such property variability so are not so sensitive to changes in nutrient uptake. Our gene expression and amino acid composition analyses alluded to the possibility of the expression of multiple spidroins in $L.\ hasselti$, as has been found in $L.\ hesperus$ [24,68,72,74]. The differential expression of a multitude of spidroins might thus be a mechanism by which cobweb building spiders maintain silk property homeostasis across variable nutritional environments.

In summary, we found that the MA silk properties of five species of Araneoid spiders varied in response to similar variations in protein intake. Stronger and tougher silks with greater crystallinity and amorphous region nanostructural alignments were found for protein fed $P.\ graeffei$ and $N.\ plumipes$ which contrasted with the findings for $A.\ keyserlingi$ and those reported previously for protein fed/deprived $N.\ pilipes$ [27]. The properties and nanostructures of $L.\ hasselti$'s MA silks were unaffected by nutrient deprivation. Our analyses suggested that variations in MaSp1: MaSp2 expression were largely ineffectual over amino acid compositions. Proline and alanine composition and the crystalline and amorphous nanostructures significantly varied in all species with the exception of $Latrodectus\ hasselti$, all with subsequent effects on mechanical properties. We uncovered additional unexpected and novel findings regarding the mechanisms inducing variations at different levels in different spiders. For instance, MaSp2 genes were not as strongly regulated as we might have predicted under the current MaSp model when protein intake changed in $A.\ keyserlingi$, $N.\ plumipes$ or $P.\ graeffei$. Rather MaSp1 was more likely to be up or downregulated in these spiders. Our modelling showed that variations in silk strength were associated with variations in crystallinity and the size, length and alignment of the crystalline and non-crystalline proteins independent of expressions of the MaSp genes. Extensibility and toughness on the other hand were driven by variations in the crystalline, amorphous and lamellar region nanostructural alignments, which were largely dis-associated from MaSp expression.

**Conclusions**

Here we holistically examined the consequences of gene expression on silk proteins and protein structure and, ultimately, silk functional properties and established that: (1) the MA silk properties of five species of Araneoid spiders varied differently in response to similar variations in protein intake, and (2) the roles of spidroin expression, crystalline, and amorphous region nanostructures on mechanical property variations differed across the species examined.
While there is broad micro-scale homogeneity in the MA silks of Araneoid spiders, our measurements found that variations in the gene expression, amino acid compositions, and nanostructures, further induce mechanical property variations between and within species. Our modelling found nanostructural variations to primarily influence silk extensibility and toughness while variations in the alignment of the crystalline and non-crystalline proteins influenced ultimate strength independent of MaSp expressions. Our study provides insights into the nanoscale mechanisms of nutritionally induced spider silk property variability by showing how spidroin expression and nanostructures affect spider silk mechanical property variations in different species. These insights further our understanding of MA silk property variability at multiple levels which is imperative if materials that match the performance of naturally spun spider silks are to one day be synthesized.

Supporting information
S1 Table. Primers used for the RT-PCR analyses, and their genbank accession numbers and sequences.
(DOCX)

S2 Table. Normalized threshold cycle (C_T) values for each of the four genes screened from the major ampullate silk glands of each of the five species (Argiope keyserlingi, Eriophora transmarina, Latrodectus hasselti, Nephila plumipes and Phonognatha graeffei) across the protein fed and protein deprived treatments.
(DOCX)

S3 Table. Means (±SE) mechanical property values. Contains statistics for five (one per species) single-factor multivariate analyses of variance.
(DOCX)

S4 Table. Means (±SE) nanostructure parameter values. Contains statistics for five (one per species) single-factor multivariate analyses of variance.
(DOCX)

S5 Table. Means (±SE) amino acid composition values. Contains statistics for five (one per species) single-factor multivariate analyses of variance.
(DOCX)

S1 Fig. Examples of WAXS derived MA silk intensity vs azimuthal angle plots at the (200) diffraction peaks.
(DOCX)

S2 Fig. Examples of WAXS derived MA silk intensity vs azimuthal angle plots at the (120) diffraction peaks.
(DOCX)

S3 Fig. Melt curves for MaSp1a and MaSp2a for the five species. Includes both protein deprived and protein fed spiders.
(DOCX)

Acknowledgments
Chen-Pan Liao assisted with the WAXS experiments and statistical analyses. Michael Kasumovic, Oliver Manlik and Kristin North provided logistical and technical assistance with the gene expression analyses. Ingi Agnarsson and anonymous reviewers provided valuable feedback on the manuscript.
Author Contributions

Conceptualization: Sean J. Blamires.
Data curation: Sean J. Blamires, Madeleine Nobbs, Penny J. Martens, I-Min Tso, Wei-Tsung Chuang, Chung-Kai Chang, Hwo-Shuenn Sheu.
Formal analysis: Sean J. Blamires, Madeleine Nobbs, Penny J. Martens, I-Min Tso, Wei-Tsung Chuang, Chung-Kai Chang, Hwo-Shuenn Sheu.
Funding acquisition: Sean J. Blamires, I-Min Tso.
Investigation: Sean J. Blamires, Madeleine Nobbs, Penny J. Martens, I-Min Tso, Wei-Tsung Chuang, Chung-Kai Chang, Hwo-Shuenn Sheu.
Methodology: Sean J. Blamires, Madeleine Nobbs, Penny J. Martens, I-Min Tso, Wei-Tsung Chuang, Chung-Kai Chang, Hwo-Shuenn Sheu.
Project administration: Sean J. Blamires.
Resources: Sean J. Blamires, Penny J. Martens, Chung-Kai Chang, Hwo-Shuenn Sheu.
Supervision: Sean J. Blamires.
Validation: Sean J. Blamires.
Visualization: Sean J. Blamires.
Writing – original draft: Sean J. Blamires.
Writing – review & editing: Sean J. Blamires, I-Min Tso, Wei-Tsung Chuang, Chung-Kai Chang, Hwo-Shuenn Sheu.

References

1. Keten S, Xu Z, Ihle M, Buehler MJ. Nanoconfinement controls stiffness, strength and mechanical toughness of β-sheet crystals in silk. Nature Mater. 2010; 9: 359–367.
2. Guan Z. Supramolecular design in biopolymers and biomimetic polymers for advanced mechanical properties. Polymer Int. 2007; 56: 467–473.
3. Stark M, Grip S, Rising A, Hedhammer M, Engstrom W, Hjalm G, et al. Macroscopic fibers self-assembled from recombinant miniature spider silk proteins. Biomacromolecules 2007; 8: 1695–1701. https://doi.org/10.1021/bm070049y PMID: 17402782
4. Tokareva O, Michalczeken-Lacerda VA, Rech EL, Kaplan DL. Recombinant DNA production of spider silk proteins. Microb Biotechnol. 2013; 6: 651–663. https://doi.org/10.1111/1751-7915.12081 PMID: 24119078
5. Alberts B, Johnson A, Lewis J, Raff M, Roberts K, Walter P. Molecular biology of the cell, 4th edition; New York, N.Y., USA: Garland Science; 2002.
6. Alsop D, Vijayan MM. Development of the corticosteroid stress axis and receptor expression in zebrafish. Reg Integr Comp Physiol. 2007; 294: R711–R719.
7. Ayoub NA, Garb JE, Tinghitiella RM, Collin MA, Hayashi CY. Blueprint for a high-performance biomaterial: full-length spider dragline silk genes. PLoS One 2007; 2: e514. https://doi.org/10.1371/journal.pone.0000514 PMID: 17565367
8. Hsia Y, Gnesia E, Jeffery F, Tang S, Viera C. Spider silk composites and applications. In: Cuppoletti J editor. Metal, Ceramic and Polymeric Composites for Various Uses. Rijeka, Croatia: Intech; 2011. pp. 303–324.
9. Chung H, Kim TY, Lee SY. Recent advances in production of recombinant spider silk proteins. Curr Op Biotechnol. 2012; 23: 957–964.
10. Babb PL, Lahens NF, Correa-Garhwal SM, Nicholson KN, Kim EJ, Hogenesch JB, et al. The Nephila clavipes genome highlights the diversity of spider silk genes and their complex expression. Nat Gen. 2017; 49: 895–903.
11. Blamires SJ, Blackledge TA, Tso IM. Physico-chemical property variation in spider silks: ecology, evolution and synthetic production. Ann Rev Entomol. 2017; 62: 443–460.

12. Hu XW, Vasanthavada K, Kohler K, McNary S, Moore AMF, Vierra CA. Molecular mechanisms of spider silk. Cell Mol Life Sci. 2006; 63: 1986–1989. https://doi.org/10.1007/s00018-006-6090-y PMID: 16819558

13. Rising A, Johansson J. Towards spinning artificial spider silk. Nat Chem Biol. 2015; 11: 309–315. https://doi.org/10.1038/nchembio.1789 PMID: 25885958

14. Vollrath F, Porter D, Holland C. There are many more lessons still to be learned from spider silk. Soft Matt. 2011; 7: 9595–9600.

15. Vollrath F, Porter D, Holland C. The science of silks. MRS Bull. 2013; 38: 73–80.

16. Larracas C, Hekman R, Dyrness S, Arata A, Williams C, Crawford T, et al. Comprehensive proteomic analysis of spider dragline silk from black widows: a recipe to build synthetic silk fibers. Int J Mol Sci. 2016; 17: 1537.

17. Xu M, Lewis RV. Structure of a protein superfiber: spider dragline silk. Proc Nat Acad Sci. USA 1990; 87: 7120–7124. PMID: 201x567

18. Parkhe AD, Seeley SK, Gardner K, Thompson L, Lewis RV. Structural studies of spider silk proteins in the fiber. J Mol Recog. 1997; 10: 1–6.

19. An B, Hinman MA, Holland GP, Yarger JL, Lewis RV. Inducing β-Sheets formation in synthetic spider silk fibers by aqueous post-spin stretching. Biomacromolecules 2011; 12: 2375–2381. https://doi.org/10.1021/bm200463e PMID: 21574576

20. Shi X, Yarger JL, Holland GP. Elucidating proline dynamics in spider dragline silk fibre using 2H–13C HETCOR MAS NMR. Chem Comm. 2014; 50: 4856–4859. https://doi.org/10.1039/c4cc00971a PMID: 24686512

21. Heim M, K eerl D, Scheibel T. Spider silk: from soluble protein to extraordinary fiber. Angew Chem Int Ed. 2009; 48: 3584–3596.

22. Lefevre T, Leclerc J, Rioux-Dube JF, Buffe teau T, Paquin MC, Rousseau ME, et al. Conformation of spider silk proteins in situ in the intact major ampullate gland and in solution. Biomacromolecules 2007; 8: 2342–2344. https://doi.org/10.1021/bm7005517 PMID: 17658884

23. Liu Y, Sp onner A, Porter D, Vollrath F. Proline and processing of spider silks. Biomacromolecules 2008; 9: 116–121. https://doi.org/10.1021/bm7005517 PMID: 18052126

24. Chaw RC, Correa-Garhwal SM, Clarke TH, Ayoub NA, Hayashi CY. Proteomic evidence for components of spider silk synthesis from black widow silk glands and fibers. J Proteom Res. 2015; 14: 4223–4231.

25. Blamires SJ, Wu CC, Wu CL, Sheu HS, Tso IM. Uncovering spider silk nanocrystalline variations that facilitate wind-induced mechanical property changes. Biomacromolecules 2013; 14: 3484–3490. https://doi.org/10.1021/bm400803z PMID: 23947397

26. Blamires SJ, Kasumovic MM, Tso IM, Martens PJ, Hook JM, Rawal A. Evidence of decoupling protein structure from spidroin expression in spider dragline silks. Int J Mol Sci. 2016; 17: 1294.

27. Blamires SJ, Liao CP, Chang CK, Chuang YC, Wu CL, Blackledge TA, et al. Mechanical performance of spider silk is robust to nutrient-mediated changes in protein composition. Biomacromolecules 2015; 16: 1218–1225. https://doi.org/10.1021/acs.biomac.5b00008 PMID: 25764227

28. Bram A, Branden CI, Craig CL, Snigireva I, Riek el C. X-ray diffraction from single fibres of spider silk. J Appl Crystallogr. 1997; 30: 390–392.

29. Rousseau ME, Lefevre T, Beaulieu L, Asakura T, Pezolet M. Study of protein conformation and orientation in silkworm and spider silk fibers using Raman microspectroscopy. Biomacromolecules 2004; 5: 2247–2257. https://doi.org/10.1021/bm049717v PMID: 15530039

30. Cetinkaya M, Xiao S, Stacklies W. Grater F. Silk fiber mechanics from multiscale force distribution analysis. Biophys J. 2011; 100: 1298–1305. https://doi.org/10.1016/j.bpj.2010.12.3712 PMID: 21354403

31. Asakura T, Suzuki Y, Nakazawa Y, Yazawa K, Holland GP, Yarger JL. Silk structure studied with nuclear magnetic resonance. Prog Nucl Magn Res Spec. 2013; 69: 23–68.

32. Holmes KC, Blow DM. The use of X-ray diffraction in the study of protein and nucleic acid structure. New York, N. Y., USA: Interscience Publishers; 1966.

33. Podorov SG, Faleev NN, Pavlov KM, Paganin DM, Stepanov SA, Forster E. A new approach to wide-angle dynamical X-ray diffraction by deformed crystals. J Appl Crystallogr. 2006; 39: 652–655.

34. Yang Z, Grubb DT, Jelinski LW. Small-angle X-ray scattering of spider dragline silk. Macromolecules 1997; 30: 8254–8261.
35. Kim S, Lee J, Jeon SM, Lee HH, Char K, Sohn BH. Orientation of lamellar nanostructures in the patterned thin films of diblock copolymer. Macromolecules 2008; 41: 3401–3404.

36. Du N, Yang Z, Liu Y, Li Y, Xu HY. Structural origin of the strain-hardening of spider silk. Adv Funct Mater. 2011; 21: 772–778.

37. Warwicker JO. Comparative studies of fibroins II. The crystal structures of various fibroins. J Mol Biol. 1960; 2: 350–362. PMID: 13783274

38. Koch MH, Bras W. Synchrotron radiation studies of non-crystalline systems. Ann Rep C Phys Chem. 2008; 104: 35–80.

39. Riekel C, Craig CL, Burhammer M, Muller M. Microstructural homogeneity of support silk spun by Eriophora fuliginea (C.L. Koch) determined by scanning X-ray microdiffraction. Naturwissenschaften 2001; 88: 67–72. PMID: 11320890

40. Sampath S, Isdebski T, Jenkins JE, Ayon JV, Henning RW, Orgel JPRO, et al. X-ray diffraction study of nanocrystalline and amorphous structure within major and minor ampullate dragline silks. Soft Matter. 2012; 8: 6713–6722.

41. Madurga R, Blackledge TA, Perea B, Riekel C, Burhammer M, Elices M, et al. Persistence and variation in microstructural design during the evolution of spider silk. Sci Rep. 2015; 5: 14820. https://doi.org/10.1038/srep14820 PMID: 26438975

42. Jenkins JE, Sampath S, Butler E, Kim J, Henning RW, Holland GP, et al. Characterizing the secondary protein structure of black widow dragline silk using solid-state NMR and X-ray diffraction. Biomacromolecules 2013; 14: 3472–3483. https://doi.org/10.1021/bm400791u PMID: 24024617

43. Breslauer DN, Lee LP, Muller SJ. Simulation of flow in the silk gland. Biomacromolecules 2009; 10: 49–57. https://doi.org/10.1021/bm800752x PMID: 19053289

44. dos Santos-Pinto JRA, Garcia AMC, Acuri HA, Esteves FG, Salles HC, Lubec G, et al. Silkomics: insight into the silk spinning process of spiders. J Proteome Res. 2016; 15: 1179–1193. https://doi.org/10.1021/acs.jproteome.5b01056 PMID: 26923066

45. Giesa T, Perry CC, Buehler MJ. Secondary structure transition and critical stress for a model of spider silk assembly. Biomacromolecules 2016; 17: 427–436. https://doi.org/10.1021/acs.biomac.5b01246 PMID: 26669270

46. Blamires SJ, Wu CL, Tso IM Variation in protein intake induces variation in spider silk expression. PLoS One 2012; 7: e31626. https://doi.org/10.1371/journal.pone.0031626 PMID: 22363691

47. Benamù M, Lacava M, Garcia LF, Santana M, Fang J, Wang X, et al. Nanostructural and mechanical property changes in spider silk as a consequence of insecticide exposure. Chemosphere 2017; 181: 241–249. https://doi.org/10.1016/j.chemosphere.2017.04.079 PMID: 28445817

48. Jeng US, Su CH, Liao KF, Chuang WT, Lai YH, Chang JW, et al. A small/wide-angle X-ray scattering instrument for structural characterization of air–liquid interfaces, thin films and bulk specimens J Appl Crystallogr. 2010; 43: 110–121.

49. Balta-Calleja FJ, Vonk CG. X-ray scattering of synthetic polymers. Polymer science library vol. 8. Amsterdam, The Netherlands: Elsevier; 1989.

50. Guinier A. X-ray diffraction in crystals, imperfect crystals and amorphous bodies; San Fransisco, CA, USA: W.H. Freeman; 1963.

51. Wutz C, Bark M, Dohrmann R, Zachmann HG Simultaneous measurements of small angle X-ray scattering, and wide angle X-ray scattering during phase transition in polymers. Rev Sci Instrum. 1995; 66: 1303–1307.

52. Glisovic A, Vehoff T, Davies RJ, Salditt T. Strain dependent structural changes of spider dragline silk. Macromolecules 2008; 41: 390–398.
58. Grubb DT, Jelinski LW. Fiber morphology of spider silk: the effects of tensile deformation. Macromolecules 1997; 30: 2860–2867.
59. Simmons A, Ray E, Jelinski LW. Solid-state $^{13}$C NMR of Nephila clavipes dragline silk establishes structure and identity of crystalline regions. Macromolecules 1994; 27: 5235–5237.
60. Fallon A, Booth RFG, Bell LD. Applications of HPLC in biochemistry. Amsterdam, The Netherlands: Elsevier; 1987.
61. Jeffrey F, La Mattina C, Tuton-Blasingame T, Hsia Y, Gnesa E, Zhao L, Franz A, Vierra C. Microdissection of black widow silk-producing glands. J Vis Exp. 2011; 47: e2382.
62. Schmittgen TD, Livak KJ. Analyzing real-time PCR data by the comparative C$_T$ method. Nat Prot. 2008; 3: 1101–1108.
63. QAIGEN. OneStep RT-PCR Handbook. Hilden, Germany: Qaigen; 2012.
64. Sturzenbaum SR, Kille P Control genes in quantitative molecular biological techniques: the variability of invariance. Comp Biochem Physiol B 2001; 130: 281–289. PMID: 11567890
65. Huggett J, Dheda K, Bustin S, Zumla A. Real-time PCR normalisation; strategies and considerations. Genes Immun. 2005; 6: 279–284. https://doi.org/10.1038/sj.gene.6364190 PMID: 15815687
66. Bustin S, Nolan T. Pitfalls of quantitative real-time reverse-transcription polymerase chain reaction. J Biomol Tech. 2004; 15: 155–166. PMID: 15331581
67. Fluidigm Corporation. Real Time PCR Users Guide. San Francisco CA, USA: Fluidigm Corporation; 2017.
68. Garb JE, Ayoub NA, Hayashi CY. Untangling spider silk evolution with spidroin terminal domains. BMC Evol Biol. 2010; 10: 243. https://doi.org/10.1186/1471-2148-10-243 PMID: 20696068
69. Gatesy J, Hayashi CY, Motriuk D, Woods J, Lewis RV Extreme diversity, conservation, and convergence of spider fibroin sequences. Science 2001; 291: 2603–2605. https://doi.org/10.1126/science.1057561 PMID: 11283372
70. Lawrence BA, Vierra CA, Moore AMF Molecular and mechanical properties of major ampullate silk of the black widow spider Latrodectus hesperus. Biomacromolecules 2004; 5: 689–695. https://doi.org/10.1021/bm0342640 PMID: 15132648
71. Garb JE, Hayashi CY. Modular evolution of egg case silk genes across orb-weaving spider superfamilies. Proc Nat Acad Sci USA 2005; 102: 11379–11384. https://doi.org/10.1073/pnas.0502473102 PMID: 16061817
72. Lane AK, Hayashi CY, Whitworth GB, Ayoub NA. Complex gene expression in the dragline silk producing glands of the Western black widow (Latrodectus hesperus). BMC Genom. 2013; 14: 846.
73. Morrison DF. Applied linear statistical methods; Englewood Cliffs, N.J., USA: Pretence Hall; 1983.
74. Ayoub NA, Hayashi CY. Multiple recombining loci encode MaSp1, the primary constituent of dragline silk, in widow spiders (Latrodectus: Theridiidae). Mol Biol Evol. 2006; 25: 277–286. https://doi.org/10.1093/molbev/msi035 PMID: 16048404
75. Blamires SJ, Chao IC, Tso IM Prey type, vibrations and handling interactively influence spider silk expression. J Exp Biol. 2010; 213: 3906–3910. https://doi.org/10.1242/jeb.046730 PMID: 21037070
76. Creager MS, Jenkins JE, Thagard-Yeaman LA, Brookes AE, Jones JA, Lewis RV. Solid-state NMR comparison of various spiders’ dragline silk fiber. Biomacromolecules 2010; 11: 2039–2043.
77. Srinivarshar y, Dida MM, Gale MD, Devos KM. Comparative analyses reveal high levels of conserved collinearity between the finger millet and rice genomes. Theor Appl Genet 2007; 115: 489–499. https://doi.org/10.1007/s00122-007-0582-5 PMID: 17619853
78. Zondlo NJ. Aromatic proline interactions: electronically tunable CH/π interactions. Acc. Chem Res. 2013; 46: 1039–1049. https://doi.org/10.1021/ar300087y PMID: 23148796
79. Nova A, Keten S, Pugno NM, Redaelli A, Buehler MJ. Molecular and nanostructural mechanisms of deformation, strength and toughness of spider silk fibrils. Nano Lett. 2010; 10: 2626–2634. https://doi.org/10.1021/nl101341w PMID: 20518518
80. Blackledge TA, Summers AP, Hayashi CY. Gumfooted lines in black widow cobwebs and the mechanical properties of spider capture silk. Zoology 2005; 108: 41–46. https://doi.org/10.1016/j.zool.2004.11.001 PMID: 16351953
81. Boutilier C, Blackledge TA. Wet webs work better; humidity, supercontraction and the performance of spider orb webs. J. Exp. Biol. 2013; 216: 3606–3610. https://doi.org/10.1242/jeb.084236 PMID: 23788700