Designs of InGaN Micro-LED Structure for Improving Quantum Efficiency at Low Current Density

Shiqiang Lu1, Jinchai Li1,2*, Kai Huang1, Guozhen Liu1, Yinghui Zhou1, Duanjun Cai1*, Rong Zhang1,2 and Junyong Kang1,2*

Abstract

Here we report a comprehensive numerical study for the operating behavior and physical mechanism of nitride micro-light-emitting-diode (micro-LED) at low current density. Analysis for the polarization effect shows that micro-LED suffers a severer quantum-confined Stark effect at low current density, which poses challenges for improving efficiency and realizing stable full-color emission. Carrier transport and matching are analyzed to determine the best operating conditions and optimize the structure design of micro-LED at low current density. It is shown that less quantum well number in the active region enhances carrier matching and radiative recombination rate, leading to higher quantum efficiency and output power. Effectiveness of the electron blocking layer (EBL) for micro-LED is discussed. By removing the EBL, the electron confinement and hole injection are found to be improved simultaneously, hence the emission of micro-LED is enhanced significantly at low current density. The recombination processes regarding Auger and Shockley–Read–Hall are investigated, and the sensitivity to defect is highlighted for micro-LED at low current density.

Keywords: InGaN, Micro-LED, Quantum efficiency, Low current density

Introduction

Group-III nitride based visible light-emitting diodes (LEDs) have a wide range of applications including signals, optical communication, information storage, backlights, displays, and general lighting (or solid-state lighting) [1, 2]. Since the first demonstration of InGaN micro-disk LED with a diameter of 12 μm by Jiang and his co-workers in 2000 [3, 4], the micro-LED has received expanding academic and industrial interests in the fields of high-resolution displays, visible light communication, bio-medicine, and sensing [5–8]. Compared with liquid–crystal display (LCD) and organic light-emitting diode (OLED), micro-LED has advantages of small size, high resolution, high luminous efficiency, high brightness, high color saturation, long operating lifetime, high response speed, and robustness, which make it the most promising candidate for the next generation display technology [9, 10]. Application scenarios of micro-LED display include high-end televisions (TV), laptops, handheld and mobile terminal devices, virtual reality (VR), augmented reality (AR), transparent display, and so on. According to Yole’s estimate, the market of micro-LED display will reach 330 million units by 2025 [11].

Traditional applications of group-III nitride LEDs, especially solid-state lighting, require the optical output power to be as high as possible [12]. In the last three decades, motivated by high-power applications, the modern research for nitride LEDs, including the design and optimization of the epitaxial structure, the study of the operating behavior and physical mechanism of the
device, and the improvement of fabrication technology, is mainly focused on the large chip, high working current density, and high input/output power [12, 13]. Foremost, the active region of nitride LED has evolved from the simplest heterostructure and single quantum well (QW) at the early stage to today’s multiple quantum wells (MQWs) with 5 periods QWs, and the QW number even can reach 8 or 10 for several commercialized high-power devices [14–17]. The EBL was proposed to block the leakage of electrons at high injection current density, even it also can impede the hole injection at some certain level [18, 19]. For conventional high-power LEDs, the most substantial issue is the reduction of the external quantum efficiency (EQE) with the increased current density, which is known as efficiency droop. The intrinsic mechanism of this behavior is related to the indirect Auger recombination at high injected carrier density [20]. As for the fabrication, thin-film flip-chip and vertical injection geometry are developed to expand the power level of nitride LEDs [21, 22].

Considerable advances have been made for the traditional large-size high-power nitride LEDs, and some of the lessons learned can be leveraged for the study of micro-LEDs. However, micro-LEDs are still far different from their high-power counterparts. The different operating behaviors, mechanisms, and conditions of micro-LED can result in different challenges and research pathways [23, 24]. For traditional LEDs, the peak EQE is located at high current density, and the working current density is high and beyond the peak efficiency (>30 A cm⁻²). But for the micro-LED emissive display, the working current density should be much lower and often in the range of 0.02 to 2 A cm⁻² [24]. At this low current density, the EQE of traditional nitride LED is very low and not sufficient for the practical micro-LED display applications. By utilizing the benefits of V-pit to screen dislocations and enhance hole injection, Zhang and his co-works have created efficient InGaN-based LEDs with peak efficiency up to 24.0% at 0.8 A/cm². However, the LED chips reported by Zhang et al. is still limited to of traditional size (1 mm × 1 mm), which is much larger than that of micro-LED [25]. Moreover, many works have reported that the internal quantum efficiency (IQE) and EQE of micro-LEDs decrease as the chip size is reduced [26–28]. This size-dependent degradation is mainly attributed to the surface recombination and the sidewall damages induced by plasma-assisted dry etching. These side-wall effects contribute to the Shockley–Read–Hall (SRH) non-radiative recombination, then decrease the quantum efficiency, and become much more serious with a smaller chip size because of the larger specific surface/side-wall area compared with the active region of the device [29–31]. To address this issue, sidewall passivation using dielectric materials and wet etching using buffered hydrofluoric acid or photoelectrochemical method were proposed to minimize these effects to a certain level for the micro-LED [31–33]. However, even improved by the sidewall passivation, the peak EQE of micro-LEDs (with a size smaller than 60 μm) is still lower than 25%, and dramatically decreased to several percent at a current density lower than 2 A cm⁻² [34, 35]. Especially for the InGaN-based red micro-LED, the currently reported EQE is quite limited to lower than 1%, due to the strong polarization and poor crystal quality [36]. Moreover, with the chip size reduced, the position of peak EQE also moves to a higher current density, which will further impede high efficiency at low current density [26].

Therefore, improving quantum efficiency at low current density becomes the large challenge and critical scientific issue for the micro-LED emissive display application. For this purpose, peak efficiency should be increased, and the onset position of efficiency should be shifted to an appropriate lower current density. Previously works mainly focused on the improvement of fabrication technology such as sidewall passivation. To improve efficiency, investigating the operating behaviors and physical mechanisms of micro-LED at low current density, which is still relatively unexplored and lack understanding, is also essential. In addition, to create device that can improve the efficiency at low current density with a maximum value, the epitaxial structure of micro-LED also needs to be re-designed and optimized, which should be quite different from their traditional large-size high-input/output counterparts. For now, the specifically designed epitaxial structure for the micro-LED emissive display operating at low current density remains lacking.

In this work, the unique challenges of micro-LED for the display application operating at low current density are highlighted, and potential solutions to address them are proposed. Using the software Advanced Physical Model of Semiconductor Devices [37], we numerically investigate the operating behaviors and physical mechanisms of InGaN micro-LEDs at various current density from 200 to 0.1 A/cm². The band diagram, wavefunction, and polarization field are simulated and analyzed for the QCSE of micro-LED, and a severer QCSE at low current density is confirmed. Influences of QW number, p-type doping concentration and AlGaN EBL on the carrier transport, carrier matching, radiative recombination and quantum efficiency of micro-LED are investigated systematically. The effect and mechanism regarding the SRH and Auger recombination are also discussed. Based on the simulation and analysis, an optimized epitaxial structure specifically designed for micro-LED operating at low current density is proposed.
Device Structures and Simulation Methods

In this study, the common structure of blue InGaN micro-LED with a rectangular chip size of 60 × 60 μm and a peak emission wavelength around 465 nm is used for the simulation. Figure 1 shows that the blue micro-LED is composed of 200 nm n-GaN layer, MQWs active region, 20 nm p-Al0.15Ga0.85 N EBL and 150 nm p-GaN layer. The MQWs active region consists of 8, 5, 3, 2 or 1 periods with 2.5-nm-thick In0.25Ga0.75 N QW embedded in 10-nm-thick In0.05Ga0.95 N quantum barrier (QB). The In composition of MQWs is adjusted and optimized to achieve the desired blue emission wavelength. The doping concentration of n-GaN, p-AlGaN EBL, and p-GaN are 2 × 10¹⁸ cm⁻³, 3 × 10¹⁸ cm⁻³, and 1 × 10¹⁹ cm⁻³, respectively.

The optical and electrical properties of micro-LEDs are numerically investigated using the software Advanced Physical Model of Semiconductor Devices [37]. In the simulation, 50% of the interface charge densities caused by spontaneous and piezoelectric polarization is assumed [38]. Except for the specifically mentioned, the SRH lifetime and Auger recombination coefficient are set as 100 ns and 1 × 10⁻³¹ cm⁶s⁻¹, respectively [39, 40]. The band offset ratio is assumed 0.7/0.3 [41], the light extraction efficiency is fixed at 70%, and the operating temperature is 300 K. Other material parameters used in the simulation can be found in Ref [42].

Results and Discussions

The Wall-plug efficiency (WPE) of LED can be expressed as follows:

\[
\text{WPE} = \frac{P_{\text{out}}}{I_{\text{op}} \times V} = \eta_{\text{EQE}} \frac{h \omega}{e \times V} = \eta_{\text{EQE}} \times \eta_{\text{elect}} \tag{1}
\]

where \(P_{\text{out}}\) denotes the optical output power, \(I_{\text{op}}\) is the operating current, \(V\) is the drive voltage, \(h\omega\) is the photon energy, and \(\eta_{\text{elect}}\) is the electrical efficiency. EQE \(\eta_{\text{EQE}}\) is a product of current injection efficiency (CIE) \(\eta_{\text{CIE}}\), IQE \(\eta_{\text{IQE}}\) and light extraction efficiency (LEE) \(\eta_{\text{LEE}}\) as described by the following equation:

\[
\eta_{\text{EQE}} = \eta_{\text{CIE}} \times \eta_{\text{IQE}} \times \eta_{\text{LEE}}. \tag{2}
\]

Furthermore, according to the ABC model [43], \(\eta_{\text{IQE}}\) can be expressed as follows:

\[
\eta_{\text{IQE}} = \frac{R_{\text{rad}}}{R_{\text{rad}} + R_{\text{SRH}} + R_{\text{Auger}}}, \tag{3}
\]

where \(R_{\text{rad}}\) is the radiative recombination rate, \(R_{\text{SRH}}\) is the SRH non-radiative recombination rate, and \(R_{\text{Auger}}\) is the Auger recombination rate. Figure 1 shows the different kinds of efficiency and the related physical mechanisms and factors.

\(R_{\text{rad}}\) and CIE must be maximized, electrical efficiency and LEE must be improved, and SRH and Auger recombination must be reduced to improve the overall efficiency of micro-LED operating at low current density. System-level approaches are required to address all these challenges. Except for light extraction, these challenges will be discussed in the following sections, and potential solutions for creating efficient epitaxial structure of micro-LED will also be proposed.
Polarization-induced QCSE is one of the dominant factors that limit the IQE of nitride LED [44]. QCSE has been widely studied for the traditional large-size high-power LED, but still lacks sufficient discussion in the context of micro-LED specific applications. Therefore, this important effect is first investigated. The active region of micro-LED discussed here is constructed by 5 period QWs, which is the most commonly used QW number for traditional nitride LED.

Figure 2a shows the energy band diagrams and the related first-level electron and hole wavefunctions of the fifth QW at 200 and 0.1 A/cm². QB and QW experience a large band bending, leading to the spatial separation of electron and hole wavefunctions. Moreover, band bending is stronger at low current density, indicating a stronger QCSE. This phenomenon is attributed to the weak screening effect with less nonequilibrium carriers at low injection current density (see Additional file 1: Fig. S1a-d, and related discussion) [41]. Figure 2b shows that a severer QCSE enhances the spatial separation of carrier wavefunctions at low current density, which leads to a lower radiative recombination rate. The radiative recombination rates and EL spectra without and with polarization effect, i.e., without and with QCSE, are further calculated to show how the electron–hole separation quantitatively reduces emission at low current density. Figure 2c, d shows that the integral intensity of radiative recombination rate and EL intensity are reduced approximately 84.0% and 72.3% by QCSE, respectively. These results indicate that it is more difficult to improve the efficiency for micro-LED than their traditional high-power cousins due to the enhanced QCSE at low current density.

Moreover, the accurate and stable control of emission wavelength at different injection current densities is critical for micro-LED display, because it determines various important parameters, such as brightness, color accuracy, and saturation. However, a severer QCSE at low current density makes this a large challenge, especially for green and red micro-LEDs that require a higher Indium composition. As a direct result of the increased band bending in InGaN QWs with a higher Indium composition and a stronger QCSE, the wavelength shift versus current density becomes more pronounced due to the screening effect. The EL spectra...
of InGaN-based red, green, and blue micro-LED with various current densities are further calculated to show how the spectra shift affects color points in the display. Additional file 1: Figure S2 shows that from 0.1 to 20 A/cm², the peak wavelengths blue shift by approximately 15.8, 6.6, and 1.7 nm for red, green, and blue micro-LED, respectively. The color points created by combining the red, green, and blue micro-LEDs are also calculated, as summarized in Additional file 1: Table S1. Figure 2e illustrates the corresponding 1931-CIE (x, y) chromaticity diagram. Clearly, the color of the emission from RGB micro-LEDs greatly changed from green to blue-green when current density increases. The 1931-CIE x value crosses from 0.1676 to 0.2084, and the 1931-CIE y value crosses from 0.3891 to 0.3106. This substantial change of point versus current density greatly deteriorates the performance of micro-LED display.

Many reports have suggested several approaches to suppress the QCSE, such as using InGaN/AlGaN ultrathin superlattice barriers [45], InGaN/AlGaN/InGaN MQWs with GaN interlayer barrier [46], wrinkled MQWs [47], and inserting a strain-relief layer before MQWs [48]. But QCSE is induced by the intrinsic properties of c-plane nitride material. Several adjustments of MQWs are not sufficient to suppress this effect completely. Recently, a decent efficiency has been achieved for non-polar nitride LED. If the cost of non-polar GaN substrate can be reduced, non-polar LED can be a ideal solution for decreasing the QCSE and realizing stable full-color emission [49].

**Carrier Transport and Matching**

The electron \( J_n \) and hole \( J_p \) current density can be expressed as follows:

\[
J_n = \sigma_n |E| = nq\mu_n |E|,
\]

\[
J_p = \sigma_p |E| = pq\mu_p |E|,
\]

where \( \sigma_n \) and \( \sigma_p \) denote the conductivity, \( n \) and \( p \) are the concentration, \( \mu_n \) and \( \mu_p \) are the mobility for the electron and hole, respectively, and \( E \) denotes the electric field. In nitride, hole concentration is an order of magnitude lower than the electron [50], and hole mobility is two orders of magnitude lower than electron [51]. This asymmetry of concentration and mobility leads to the mis-matching of carrier flux \( J_n > J_p \) and deteriorates the performance of LED in following two ways.

**Current injection efficiency:** Current injection efficiency \( \eta_{\text{CIE}} \) represents the ratio of the recombination current \( J_{\text{recomb}} \) to the total current \( J_{\text{total}} \), as following equation [52]:

\[
\eta_{\text{CIE}} = \frac{J_{\text{recomb}}}{J_{\text{total}}} = \frac{J_n - \text{recomb} + J_p - \text{recomb}}{J_n + J_p} = \frac{2 \times J_p}{J_n + J_p}.
\]

\( J_{\text{recomb}} \) depends on the smaller one of carrier currents, i.e., hole current. Equation (6) confirms that the carrier mis-matching \( J_n > J_p \) limits the maximization of CIE. Radiative recombination rate: The radiative recombination rate \( R_{\text{rad}} \) is described by Fermi’s golden rule as the following equation:

\[
R_{\text{rad}} = C \times f \frac{dE_{\text{cv}} \hbar v_{\text{cv}} |M_{T}(E_{\text{cv}})|^2 \rho_{e}(E_{\text{cv}}) f_{e}(1 - f_{e})}{E_{\text{cv}} - E_{\text{cv}}},
\]

where \( C \) is a constant, \( E_{\text{cv}} \) is the transition energy, \( \hbar \) is the Planck constant, \( v_{\text{cv}} \) is the frequency of the generated light, \( \rho_{e} \) is the reduced density of states, \( f_{e} \) and \( f_{e} \) are the Fermi–Dirac distributions, and \( |M_{T}|^2 \) is the momentum matrix element [53]. Less hole and additional electron accumulation in the QW can lead to the expansion of crystal lattice and the buildup of tensile strain. Under this stress variation, the charge densities of quantum levels around the valence band maximum are reduced. This further decreases the optical transition probability and reduces \( R_{\text{rad}} \) according to Eq. (7). In this manner, the local carrier mis-matching in a single QW also limits the IQE. More specific discussion about this topic can be found in previous reports [54–56].

In the following sections, the influences of QW number, \( p \)-type doping concentration, and EBL structure on the carrier transport will be analyzed to determine the best carrier matching conditions. Finally, an optimized epitaxial structure for the efficient micro-LED display operating at low current density will be proposed.

**Carrier Mis-Matching in 5QWs**

First, the carrier transport properties of blue micro-LED with 5QWs are simulated. The distributions of carrier concentration at 200 A/cm² and 0.1 A/cm² are illustrated in Additional file 1: Fig. S3a and b, respectively. The inhomogeneous distribution in 5 QWs can be observed both at high and low current densities. Additional file 1: Figure S3c and d shows that the electron mobility in MQWs (684 cm²V⁻¹s⁻¹) is two orders of magnitude higher than the hole mobility (10 cm²V⁻¹s⁻¹). Hence, electrons may just inject into, then pass through MQWs without participating in the recombination, leading to the electron leakage problem and a low CIE [51].

Figure 3a shows the distribution of electron and hole current density at 200 A/cm². Total hole current density (217.4 A/cm²) is only 65.2% of the electron (333.3 A/cm²), indicating a severe mis-matching of carrier and a low CIE. The leakage electron current is as high
as 116.0 A/cm², which deteriorates both the radiative efficiency and the hole injection. As shown in Fig. 3b, the leakage electron current is only 0.01 A/cm², and the calculated η_{CIE} is as high as 95% at 0.1 A/cm². These results indicate that achieving a high CIE is easier at low current density. However, except the 5th QW where \( J_p \) can be equal to \( J_n \), the carrier mis-matching and additional electron accumulation are quite severe in other four QWs (QW 1, 2, 3 and 4) both at high and low current density. At 200 A/cm², the electron current densities of these four QWs are 120, 43, 16 and 5 times higher than the hole current density (Fig. 3a). At 0.1 A/cm², they are 23, 9, 4, and 2 times higher than the hole current (Fig. 3b). Based on Eq. (7), this great carrier mis-matching evidently decreases the radiative recombination rate of these four QWs. Therefore, radiative recombination rates in these four QWs are about 3.4%, 4.0%, 10.1%, and 34.2% at 200 A/cm², and 11.3%, 10.1%, 10.7% and 21.2% at 0.1 A/cm² compared with the 5th QW. These carrier mis-matching and low radiative emission finally reduce the monolithic efficiency of micro-LED.

Factors Influencing Carrier Transport and Matching

QW number: For traditional large-size LED operating at high current density, MQWs with 5, even 8 or 10 periods must be used to achieve a high optical output power. However, for the micro-LED emissive display, the output power is much smaller and the working current density is much lower. As discussed in the previous section, even at low current density, the carrier mis-matching remains quite severe in 5QWs, and only one QW can achieve the best matching condition. From this perspective, for the micro-LED operating at low current density, the active region with less QW number should be a better design for improving the efficiency due to the improved carrier matching.

The effect of the QW number on the micro-LED is investigated to verify our assumption. Figure 4a–f shows the carrier current density and radiative recombination rate at 0.1 A/cm² of LEDs with 3QWs, 2QWs, and 1QW, respectively. The current curves have only one intersection point (one point of best carrier matching, \( J_p = J_n \)) due to the monotonically decreasing tendency of current, but with fewer QWs, as the cases of 3QWs and 2QWs, two intersection points can be achieved in two different QWs.
(Fig. 4a, b). In other word, the carrier mis-matching in MQWs can be partially overcome with fewer QWs. Especially for the 2QWs, with appropriate adjustment, the perfect matching of carrier flux can be achieved in all two QWs. Radiative recombination rate is also higher in the 2QWs than 3QWs and 5QWs because the carrier consumption by the radiative recombination is more concentrated in the active region with fewer QWs (Figs. 3d, 4d, e). Undoubtedly, the best carrier matching is in the LED with only one QW, and the radiative recombination rate is also highest for the 1QW, as shown in Fig. 4c, f.

Figure 5a, b shows IQE as a function of the current density between 0 to 200 A/cm² and 0 to 10 A/cm², respectively. For a current density higher than 50 A/cm², IQE decreases when the QW number is reduced. By contrast, IQE with a current density lower than about 30 A/
cm² increases when QW number decreases. At 0.1 A/cm², IQE values for 8, 5, 3, 2, and 1 QWs are 55%, 62%, 69%, 77%, and 78%, respectively. Moreover, as shown in Fig. 5b, the position of peak IQE also moves from 6.0 A/cm² in 8 QWs to the lowest current density approximately 1.2 A/cm² in 2QWs. The IQE curves at low current density (< 1 A/cm²) also become steeper and sharper with a lower QW number, indicating that achieving the highest efficiency is easier and quicker. This is quite beneficial for improving the efficiency at low current density. The physical mechanism behind this tendency of IQE can be explained by the better matching of carrier flux, and more concentrated, stronger radiative emission in the active region with fewer QW number. As shown in Fig. 5c, compared with 8 QWs, the integral EL intensity of 5, 3, 2, and 1 QWs at 0.1 A/cm² are increased about 6.1%, 14.8%, 28.4%, and 32.1%, respectively. This result confirms that not only the efficiency but also the output power is improved with fewer QW number.

The simulation results show that 1QW has the highest EL intensity and maybe the best structure for micro-LED operating at low current density. However, experimentally, it is difficult to epitaxially grow just one QW which has flat surface and sharp interface. Moreover, for the 1QW structure, the position of peak IQE increases slightly to 2.9 A/cm², and the shape of IQE curve also slightly changed. This can be explained by the special circumstance of the single QW. Compared with other QWs, the QW adjoined to the EBL has a special polarization environment and it is considered as a “problem QW.” This topic will be discussed with more details in the section of AlGaN EBL. Considering these reasons, the 2QWs should be the best active region design, which has the similar good matching of carrier flux, close IQE value, and EL intensity to IQW. Therefore, in the following sections, all simulations are based on the micro-LED with 2QWs.

P-GaN doping concentration: The performance of 2QWs LEDs with different p-type doping concentrations in p-GaN is further investigated. As shown in Fig. 6a, when the doping concentration of p-GaN increasing from $1 \times 10^{18}$ cm⁻³ to $5 \times 10^{19}$ cm⁻³, the radiative recombination rates at 0.1 A/cm² further increases about 3.1% and 3.0% for the two QWs. Figure 6b shows that the
total hole current density increases from 0.157 to 0.162 A/cm², meanwhile, leakage electron current density is reduced from 0.009 to 0.005 A/cm² with the increase of doping concentration. It's worth noting that the recombination current in the QW near n-side is higher than the QW near p-side (Fig. 6b). Therefore, the radiative recombination rate near the n-side QW is also slightly higher than that near the p-side QW. As shown in Fig. 6c, one can find that the energy barrier for carriers in EBL, which is defined as the energy distance between the electron/hole quasi-Fermi level and the highest conduction band or lowest valence band, are almost unchanged under different doping concentration of p-GaN, that is, the hole injection is not improved by increasing doping concentration. Figure 6d shows the average hole concentration in the p-GaN and QWs. The hole concentration in the p-GaN is almost exponentially dependent on the doping concentration. Inside the QWs, the hole concentration is increased approximately twice from 1.59 × 10¹⁹ cm⁻³ to 3.08 × 10¹⁹ cm⁻³ with a higher doping concentration. These results indicate that the increased hole concentration is the main contribution for the improvement of radiative recombination. Therefore, even at low current density, the p-type doping problem of nitride remains notable, and enhancing the doping efficiency and hole concentration is still beneficial for the efficiency of micro-LED.

**AlGaN EBL**: In the last decades, a 10–20 nm p-type AlGaN EBL has become a standard structure for traditional nitride-based LEDs. This EBL is considered to block the electron leakage and suppress the efficiency droop under high injected current density. In spite of this, it is worth noting that the EBL is still a very complicated, subtle structure for the LED. It involves many important factors, including Al composition, p-type doping efficiency of AlGaN, band offset ratios, and polarization effect. Each of them can influence the band structure and carrier transport substantially, then determines the effectiveness of EBL. For the construction of EBL, thickness, composition, and doping concentration must be considered and optimized carefully to balance the enhancement of electron confinement and the blocking of hole injection, otherwise, the opposite may happen, and the performance of LED might deteriorate. For micro-LED, the effectiveness of EBL for operating at low current density must be reconsidered, which may be different with the case of traditional high input/output LED.

1. **Doping concentration of EBL**: First, the effect of EBL doping concentration on carrier transport at low current density is investigated. The thickness and Al composition of EBL are fixed as 20 nm and 0.15, respectively. Considering the low solubility of Mg dopant in AlGaN, the crystal degradation, and compensation effect by over-doping [50], the doping concentration of EBL is first set as be 3 × 10¹⁸ cm⁻³. Figure 7a shows the corresponding energy band structure. Clearly, despite the existence of EBL, the electron leakage out of the QW still can be caused by the insufficient electron confinement due to the downward bending of the last QW and EBL. A new energy valley under the electron quasi-Fermi level appears at the interface of last QB and EBL. Thus, electrons would escape from QW and accumulate in this area. This distortion of band structure makes the EBL relatively ineffective, and it can be contributed to the polarization effect. As shown in Fig. 7c, the strong polarization induces a large amount of charges at the interfaces. Due to the unbalanced polarization charges are positive at the interface of the last QB/EBL, a large electrostatic field pointing from the p-side to the n-side builds up in the last QB, which is opposite to the fields in other QBs and EBL. These electrostatic fields pull down the energy band of the last QB and EBL. Moreover, the electric fields in the last QW and last QB both can attract electrons and drive them out of the active region into the p-layer. This can be observed in the carrier concentration diagram, as shown in Fig. 7d. The dotted black line indicates that a part of the electrons escape from the active region and accumulate at the interface of last QB/EBL. In the EBL and p-GaN, the leakage electron remains relatively high.

On the other hand, the EBL also introduces a potential barrier as high as 480 meV for hole injection. Moreover, as shown in Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole quasi-Fermi level appears at the interface region between the EBL/p-GaN due to the polarization effect. As indicated by the solid black line of Fig. 7a, c, an energy valley close to the hole qua...
in actual experiment, $6 \times 10^{19}$ cm$^{-3}$ is selected as a new doping concentration in the EBL. As shown in Fig. 7b, by increasing the doping concentration, the valence band of EBL is lifted due to the alignment of hole quasi-Fermi level, resulting in a reduced hole energy barrier of 281 meV. Moreover, the high $p$-type doping also helps lower the electron quasi-Fermi level with respect to the conduction band in EBL, hence increasing the effect barrier for electron leakage to 1175 meV. These changes improve electron confinement and hole injection. Figure 7d shows that compared with the doping concentration of $3 \times 10^{18}$ cm$^{-3}$, the hole concentration in the active region is greatly increased, and the leakage electron in the EBL and $p$-GaN is reduced to almost zero. However, the energy valley at the interface between the EBL/$p$-GaN still exists. Moreover, the upward of valence band also introduces a new energy valley for the hole accumulation at the interface between the last QB/EBL, which can be confirmed by the hole concentration distribution in Fig. 7d. These energy valleys can impede the hole injection into QWs, hence compensating the advantage of high doping concentration.

**b. Al composition of EBL:** Compared with the increase of doping concentration, reducing the composition of AlGaN EBL may be an easier, more efficient method to improve the carrier transport at low current density. The effectiveness of EBL is sensitively dependent on Al composition, band offset, and polarization effect. Increasing the Al composition of EBL can increase the band offset between the last QB/EBL, which increases the electron barrier height. However, as shown in Fig. 8a, the polarization-induced charges at the interfaces also increase accordingly, which pull down the electron barrier height. Two mechanisms have the opposite effect for confining electrons.

Based on this analysis, the carrier transport of 2QWs micro-LEDs with different EBL structures at 200 and 0.1 A/cm$^2$ are investigated. The result of effective energy barrier heights of different EBL are summarized in Table 1. First of all, both the electron and hole energy barriers at

---

**Fig. 7** Energy band diagrams of 2QWs LED with (a) $3 \times 10^{18}$ cm$^{-3}$ and (b) $6 \times 10^{19}$ cm$^{-3}$ doping concentration in AlGaN EBL. (c) Space charge density and (d) carrier concentration distribution of 2QWs LED with $3 \times 10^{18}$ and $6 \times 10^{19}$ cm$^{-3}$ doping concentration of EBL. The inset of (c) shows the direction of polarization fields.
low current density are much higher than the cases of at high current density due to the lower nonequilibrium carrier population. A higher energy barrier can improve the electron confinement, but also severely impedes the hole injection at low current density. This indicates that the influence of EBL on the carrier transport of micro-LED operating at low current density is much higher than that of the traditional high input/output LED. Table 1 also shows that with a relatively low Al composition. By further increasing the composition, the electron barrier increases along with composition, meaning that the band offset becomes dominant over the polarization effect. Compared with high current density, this turning point of composition is higher at low current density due to the enhanced polarization effect by less carrier screening. On the other hand, because of the original band offset between the last QB/p-GaN and the band bending, there are energy barriers for electron and holes exist at the interface even without the EBL. At 200 A/cm², when the composition is lower than 0.20, the electron energy barrier is lower than the case of without EBL, but the hole barrier is higher approximately 64 meV. At 0.1 A/cm², even with composition higher than 0.20, the electron barrier of AlGaN EBL (523 meV) is still 151 meV lower than the case of without EBL (674 meV), but the hole barrier is increased approximately 76 meV from 409 to 485 meV. These results indicate that both the electron confinement and hole injection could be deteriorated by the EBL with an incorrect composition, especially for micro-LED operating at low current density.

For a deep analysis, band structures of micro-LEDs with Al$_{0.20}$Ga$_{0.80}$ N EBL and without EBL as representatives are illustrated in Fig. 8b, c. The EBL introduces two energy valleys at the interface of last QB/EBL and EBL/p-GaN for electron and hole accumulation, respectively, which can be confirmed by the carrier concentration diagram in Fig. 8d. Therefore, the electron confinement and
hole injection are poor for this structure. When the EBL is removed, as shown in Fig. 8c, the energy barrier for electron is increased, and the energy valley for electron extracting and accumulation disappears. These changes prevent electrons leakage more effectively, as confirmed in Fig. 8d. Meanwhile, the barrier height for hole injection is reduced, and the energy valley at the EBL/p-GaN interface is also removed. So, the hole can transport directly into the QW without facing large obstacle, as shown in Fig. 8c, d.

The above careful investigation suggests that without EBL may be a better structure for the micro-LED operating at low current density. Simulation results support our suggestion. Figure 8e illustrates the carrier current density at 0.1 A/cm² with different EBL structures. When the Al composition of the EBL is reduced from 0.15 to 0.04, the total electron and hole current densities increase from 0.167 and 0.159 A/cm² to 0.225 and 0.225 A/cm², respectively. Moreover, when the EBL is completely removed, both the total electron and total hole current densities greatly increase to 0.528 A/cm², which is approximately 3 times higher compared with the Al₀.₁₅Ga₀.₈₅ N EBL. This enhancement is contributed to the improved electron confinement and hole injection.

Figure 8f shows the IQE curves at low current density. When the Al composition of EBL is reduced from 0.15 to 0.04, the IQE values increase evidently due to the improved carrier transport. However, by removing the EBL, the IQE value experiences a slight decrease compared with Al₀.₀₄Ga₀.₉₆ N EBL. This can be explained by the carrier matching in two QWs. As indicated by the green arrows in Fig. 8e, a, perfect carrier matching occurred in both the two QWs with Al₀.₀₄Ga₀.₉₆ N EBL. With the increase of current density by removing EBL, the matching of electron and hole flux has been slightly broken in the first QW, where the electron current density is slightly higher than the hole. Therefore, the IQE is slightly reduced because of this carrier mis-matching in one QW.

The superiority of micro-LED without EBL is still remarkable due to the improved carrier transport. As shown in Fig. 8g, at 0.1 A/cm², the integral EL intensities of micro-LED without EBL are 3.53, 3.23, and 2.38 times higher compared with the LED with Al₀.₁₅Ga₀.₈₅ N, Al₀.₁₀Ga₀.₉₀ N and Al₀.₀₅Ga₀.₉₆ N EBL, respectively. Moreover, as shown in the inset of Fig. 8g, the working voltages under 1 A/cm² and 0.1 A/cm² are reduced about 0.53 V and 0.57 V by removing the EBL, respectively. This improves the electrical efficiency, then finally increases the WPE of micro-LED. To further confirm that the EBL-free structure is a better design for micro-LED operating at low current density, another simulation is performed using the reported blue micro-LED structure with maximal known efficiency. The results and discussions can be found in the Supporting Materials (Additional file 1: Fig. S4a-d).

**Optimized Structure for Micro-LED Operating at Low Current Density**

Based on above simulation and analysis, the optimized epitaxial structure specifically designed for the efficient micro-LED emissive display operating at low current density is proposed, as shown in Fig. 9. Three principles must be followed. First, in contrast to the traditional large-size high-power nitride LED, the QW number of micro-LED should be reduced to just two, which has a better condition for the carrier matching, a more concentrated radiative emission, and higher IQE and WPE. Second, the p-type doing still needs to be enhanced due to the relatively low hole concentration and mobility compared with the electron in nitride, which demands a more efficient p-type doping strategy. Third, to improve the carrier transport and matching, the doping concentration of AlGaN EBL should be greatly enhanced, or the AlGaN EBL can be completely removed. Without using the AlGaN EBL, the electron confinement, hole injection, carrier matching, IQE, and WPE of the micro-LED can be greatly improved at low current density.

**Auger Recombination and SRH Non-radiative Recombination**

Based on Eq. (3), except for radiative recombination, the SRH and Auger recombination also play critical roles in the IQE of LED. Hence, it is important to investigate the effects and mechanism regarding the SRH and Auger recombination for the micro-LED. In this part, the LED structure with 2QWs is still used, and all the simulation parameters are the same as mentioned in the Methods except for SRH lifetimes.

**Negligible Auger and Prominent SRH Recombination**

The SRH recombination rate can be expressed as follows [57]:

\[
R_{SRH} = \frac{np - n_i^2}{\tau_p \left( n + n_i \exp \left( \frac{E_T}{kT} \right) \right) + \tau_n \left( p + n_i \exp \left( \frac{E_T}{kT} \right) \right)}.
\]  

\[
\tau_p = \frac{1}{c_p N_t}, \quad \tau_n = \frac{1}{c_n N_t},
\]

\[
c_p = \sigma_p v_p, \quad c_n = \sigma_n v_n,
\]

where \( n_i \) is the intrinsic carrier concentration, \( \tau_p \) and \( \tau_n \) are the hole and electron SRH lifetimes, respectively, \( E_T \) is the energy difference between the trap level and the trap level.
intrinsic Fermi level, \( c_p \) and \( c_n \) are the capture coefficients for electron and hole, \( N_t \) is the trap density, \( \sigma_p \) and \( \sigma_n \) are capture cross sections for electron and hole, and \( \nu_p \) and \( \nu_n \) are the average thermal velocities of electron and hole, respectively. According to Eqs. (8)-(10), the SRH recombination of a trap is completely specified by its density, capture cross sections and energy level.

The Auger recombination rate is given by the following:

\[
R_{\text{Auger}} = (C_n n + C_p p)(np - n_i^2),
\]

where \( C_n \) and \( C_p \) are the Auger recombination coefficients.

Given that the injected hole and electron concentrations are much higher than the intrinsic carrier concentration in the undoped QWs (according to simulation result, the highest carrier concentration in QW is only approximately \( 10^7 \) cm\(^{-3} \) in the absence of externally injected current), the SRH and Auger recombination rate can be further simplified as the following equations:

\[
R_{\text{SRH}} = \frac{np}{\tau_p n + \tau_n p},
\]

\[
R_{\text{Auger}} = (C_n n + C_p p)np.
\]

Equations (12) and (13) clearly show that \( R_{\text{SRH}} \) is in direct proportion to the first power of the carrier concentration, that is, \( R_{\text{SRH}} \) is sensitive to low current density, while the \( R_{\text{Auger}} \) is more dominant at high current density.

This theoretical analysis agrees with our simulation results. Figure 10a, b shows the calculated radiative, SRH, and Auger recombination rates at 200 and 0.1 A/cm\(^2\), respectively. At high current density, the Auger recombination rate (about \( 0.8-1.4 \times 10^{29} \) cm\(^{-3} \) s\(^{-1} \)) is comparable with the radiative rate (about \( 4.2-6.0 \times 10^{29} \) cm\(^{-3} \) s\(^{-1} \)). In fact, the substantial problem of efficiency droop at high drive currents is now widely acknowledged as caused by the Auger recombination [20]. While, at low current density, relatively, the Auger recombination rate dramatically decreases to two orders of magnitude lower (about \( 6.3-7.2 \times 10^{22} \) cm\(^{-3} \) s\(^{-1} \)) than the radiative recombination (about \( 3.7-4.0 \times 10^{24} \) cm\(^{-3} \) s\(^{-1} \)). Therefore, the Auger recombination should be negligible at low current density. Conversely, with the decrease of current density, the SRH recombination rate relatively increases from a small value at 200 A/cm\(^2\) (two orders of magnitude lower than radiative recombination) to a level comparable with the radiative emission at 0.1 A/cm\(^2\). As a result, the micro-LED operating at low current density requires improvement in the SRH or defect recombination instead of the Auger recombination.

Requirement for Low Defect Density

According to Eq. (9), SRH lifetimes, \( \tau_p \) and \( \tau_n \), are in inverse proportion to the density of defects \( N_t \). Therefore, the effect of defect density can be estimated by simply changing the SRH lifetime in the simulation. Figure 10c
shows the calculated ratio of SRH/radiative recombination rate at various SRH lifetimes. With the decrease of SRH lifetimes from 150 to 50 ns, i.e., the increase of defect density, the SRH/radiative ratio slightly increases from 0.01 to 0.03 at 200 A/cm², but greatly increases from 0.15 to 0.43 at 0.1 A/cm². This means that a much larger percentage of carriers is consumed by the trapping defects at low current density. Therefore the efficiency is much more sensitive to the defect density at low current than high current density. The IQE results as shown in Fig. 10d, e confirm this trend. With the decrease of SRH lifetimes from 150 to 50 ns, IQE only decreases about 0.01 at 200 A/cm², but dramatically decreases about 0.17 at 0.1 A/cm². Moreover, as shown in Fig. 10f, with the decrease of SRH lifetime, the position of peak IQE also moves from 3.1 A/cm² to a higher current density of 9.0 A/cm², and the IQE curves become less steep and sharp, which means that the threshold/onset current is increased. This is disadvantageous for improving the efficiency of micro-LED at low current density.

Compared with the traditional large-size high-power LED working at high current density, the micro-LED operating at low current density is much more sensitive to defect density, and minimizing the defect recombination is of paramount importance for achieving a high efficiency. Therefore, the micro-LED requires a much more higher crystal quality of materials than the traditional LED, and poses large challenges for the epitaxial growth of the material and the fabrication of the device for the community.

**Conclusions**

In summary, the operating behaviors, mechanisms and conditions of InGaN micro-LED operating at low current density are numerically investigated, and an optimized epitaxial structure specifically designed for the micro-LED display is proposed. Analysis of the polarization effect shows that micro-LED suffers a severer QCSE at low current density. Hence, improving the efficiency and controlling the emission color point are more difficult. Carrier transport and matching are analyzed to determine the operating conditions of micro-LED. It is shown that less QW number can improves the carrier matching and leads to higher efficiency and output power at low current density. Effectiveness of the EBL for micro-LED is analyzed, and electron confinement and hole injection are found to be improved simultaneously at low current density by removing the EBL. Moreover, simulation has shown that the Auger recombination is negligible, but the SRH recombination greatly influences the efficiency of micro-LED at low current density, which has raised higher requirements for the crystal quality of materials.
and the fabrication process of devices. Our numerical study can provide valuable guidance for creating efficient micro-LED display and promote future research in this area.

Abbreviations

Micro-LED: Micro-light-emitting-diode; QCSE: Quantum-confined Stark effect; EBL: Electron blocking layer; SRH: Shockley–Read–Hall; LCD: Liquid–crystal display; OLED: Organic light-emitting diode; TV: Televisions; VR: Virtual reality; AR: Augmented reality; QW: Quantum well; MQWs: Multiple quantum wells; EQE: External quantum efficiency; IQE: Internal quantum efficiency; QB: Quantum barrier; WPE: Wall-plug efficiency; CIE: Current injection efficiency; LEE: Light extraction efficiency.

Supplementary Information

The online version contains supplementary material available at https://doi.org/10.1186/s11671-021-03557-4.

Additional file 1. Figure S1. Space charge, electric field, transition energy and normalized EL spectra of micro-LED with SQW5. Figure S2. EL spectra of InGaN-based red, green, and blue micro-LEDs at 0.1 and 200 A/cm². Table S1. The 1931-CIE (x, y) color points created by combining the red, green, and blue micro-LEDs. Figure S3. Carrier concentration and mobility of micro-LED with SQW5. Figure S4. Simulation results from the micro-LED structure with maximal known efficiency.

Acknowledgements

Not applicable.

Authors’ contributions

SL and JL designed and made the simulation, co-wrote the manuscript. KH, GL, andnd YZ checked the simulation data and provided valuable discussion. DC, RZ, and JK co-wrote the manuscript. All authors read and approved the final manuscript.

Funding

This work was supported by National Key R&D Program of China (2018YFB0406702), NSFC (61874090, 62074133, 61974124, and 61874091) of China, Key scientific and technological Program of Xiamen (3502ZZ20191016 and 3502CQ20191001).

Availability of data and materials

The data and the analysis in the current work are available from the corresponding authors on reasonable request.

Declarations

Competing interests

The authors declare that they have no competing interests.

Author details

1 Fujian Key Laboratory of Semiconductor Materials and Applications, CI Center for OSED, College of Physical Science and Technology, Xiamen University, Xiamen 361005, People’s Republic of China. 2 Future Display Institute of Xiamen, Xiamen 361005, People’s Republic of China.

Received: 20 April 2021 Accepted: 24 May 2021

Published online: 03 June 2021

References

1. Ponce FA, Bour DP (1997) Nitride-based semiconductors for blue and green light-emitting devices. Nature 386:351–359

2. Jain SC, Willander M, Narayan J, Van OR (2000) III-nitrides: Growth, characterization, and properties. J Appl Phys 87:965–1006

3. Jin SX, Li J, Li JZ et al (2003) GaN microdisk light-emitting diodes. Appl Phys Lett 76:631–633

4. Jiang HX, Jin SX, Li J et al (2001) III-nitride blue microdisplays. Appl Phys Lett 78:1303–1305

5. Green RP, McKendry JID, Massobure D et al (2013) Modulation bandwidth studies of recombination processes in blue and green InGaN quantum well micro-light-emitting diodes. Appl Phys Lett 102:091103

6. Jeonrett AH, Neale SL, Massobure D et al (2014) Optoelectronic tweezers system for single cell manipulation and fluorescence imaging of live immune cells. Opt Express 22:1272

7. Ding K, Avrutin V, Izyumskaya N et al (2019) Micro-LEDs, a manufacturability perspective. Appl Sci 9:1206

8. Fan ZY, Lin JY, Jiang HX (2008) III-nitride micro-emitter arrays: development and applications. J Phys D Appl Phys 41:094001

9. Huang Y, Tan G, Gou F et al (2019) Prospects and challenges of mini-LED and micro-LED displays. J Soc Inf Dis 27:387–401

10. Wu T, Sher C-W, Lin Y et al (2018) Mini-LED and micro-LED: promising candidates for the next generation display technology. Appl Sci 8:18557

11. (2017) Yole Development, Micro LED displays could disrupt LCD and OLED. http://www.yole.fr/MicroLEDDisplays_Market.aspx?X9bBSkZaUn

12. Steranka FM, Bhat J, Collins D et al (2002) High power LEDs—technology status and market applications. Phys Status Solidi 194:380–388

13. Chen G, Graven M, Kim A et al (2008) Performance of high-power III-nitride light emitting diodes. Phys Status Solidi 205:1086–1092

14. Lee S-J, Kim KH, Ju J-W et al (2011) High-brightness GaN-based light-emitting diodes on Si using wafer bonding technology. Appl Phys Express 4:066601

15. Nakamura S, Krames MR (2013) History of Gallium–Nitr-based light-emitting diodes for illumination. Proc IEEE 101:2211–2220

16. Nakamura S, Senoh M, Iwasa N, Nagahama S (1995) High-power InGaG single-quantum-well structure blue and violet light-emitting diodes. Appl Phys Lett 67:1868–1870

17. Nakamura S, Senoh M, Mukai T (1993) P-GaInN/InGaN/N-GaN double-heterostructure blue-light-emitting diodes. Jpn J Appl Phys 32:L-11

18. Han S-H, Lee D-Y, Lee S-J et al (2009) Effect of electron blocking layer on efficiency droop in InGaN/GaN multiple quantum well light-emitting diodes. Appl Phys Lett 94:231123

19. Ru-Chin Tu, Tun C-J, Pan S-M et al (2003) Improvement of near-ultraviolet InGaN-GaN light-emitting diodes with an AlGaN electron-blocking layer grown at low temperature. IEEE Photonics Technol Lett 15:1342–1344

20. Kioukapis E, Rinke P, Delaney KT, Van de Walle CG (2011) Indirect Auger recombination as a cause of efficiency droop in nitride light-emitting diodes. Appl Phys Lett 98:161107

21. Shchekin OB, Epler JE, Trottier TA et al (2006) High-performance thin-film InGaN–GaN light-emitting diodes on Si using wafer bonding technology. Appl Phys Lett 89:071109

22. Kim H, Kim K-K, Choi K-K et al (2007) Design of high-efficiency GaN-based light-emitting diodes with vertical injection geometry. Appl Phys Lett 91:023510

23. Wiener JJ, Tansu N (2019) III-Nitride micro-LEDs for efficient emissive displays. Laser Photon Rev 13:1900141

24. Xia J, Zhou Y, Liu B et al (2019) A simulation study on the enhancement of the efficiency of GaN-based blue micro-LEDs at low current density for micro-LED applications. Mater Res Express 6:105915

25. Zhang S, Zhang J, Gao J et al (2020) Efficient emission of InGaN-based light-emitting diodes: toward orange and red. Photonics Res 8:1671

26. Olivier F, Tinco T, Dupré L et al (2011) Influence of size-reduction on the performances of GaN-based micro-LEDs for display application. J Lumin 191:112–116

27. Hwang D, Mughal A, Pynn CD et al (2017) Sustained high external quantum efficiency in ultrasmall blue III–nitride micro-LEDs. Appl Phys Express 10:032101

28. Tian P, McKendry JJD, Gong Z et al (2012) Size-dependent efficiency and performance of GaN-based LEDs: a size effect study. Appl Phys Lett 101:231110

29. Olivier F, Daami A, Licitra C, Templier F (2017) Shockley–Read–Hall and Auger non-radiative recombination in GaN-based LEDs: A size effect study. Appl Phys Lett 111:025204

30. Olivier F, Daami A, Licitra C, Templier F (2017) Shockley–Read–Hall and Auger non-radiative recombination in GaN-based LEDs: A size effect study. Appl Phys Lett 111:025204
30. Konoplev SS, Bulashevich KA, Karpov SY (2018) From large-size to micro-LEDs: scaling trends revealed by modeling. Phys Status Solidi 215:1700508
31. Kou J, Shen C-C, Shao H et al (2019) Impact of the surface recombination on InGaN/GaN-based blue micro-light emitting diodes. Opt Express 27:A643
32. Wong MS, Hwang D, Alhassan Al et al (2018) High efficiency of III-nitride micro-light-emitting diodes by sidewall passivation using atomic layer deposition. Opt Express 26:21324
33. Chan L, Karmstrand T, Chan A et al (2020) Fabrication and chemical lift-off of sub-micron scale III-nitride LED structures. Opt Express 28:35038
34. Wong MS, Lee C, Myers DJ et al (2019) Size-independent peak efficiency of III-nitride micro-light-emitting diodes using chemical treatment and sidewall passivation. Appl Phys Express 12:097004
35. Hang S, Chuang C-M, Zhang Y et al (2021) A review on the low external quantum efficiency and the remedies for GaN-based micro-LEDs. J Phys D Appl Phys 54:153002
36. Pasayat SS, Gupta C, Wong MS et al (2021) Demonstration of ultra-small (<10 μm) InGaN micro-LEDs with useful on-wafer external quantum efficiency (>0.2%) for mini-displays. Appl Phys Express 14:011004
37. Software C (2020) APSYS–Crosslight Software. http:// www.crosslight.com/products/apsys/
38. Fiorentini V, Bernardini F, Ambacher O (2002) Evidence for nonlinear macroscopic polarization in III–V nitride alloy heterostructures. Appl Phys Lett 80:1204–1206
39. Usman M, Saba K, Han D-P, Muhammad N (2018) Efficiency improvement of green light-emitting diodes by employing all-quaternary active region and electron-blocking layer. Superlattices Microstruct 113:585–591
40. Piprek J, Römer F, Witzgmann B (2015) On the uncertainty of the Auger recombination coefficient extracted from InGaN/GaN light-emitting diode efficiency droop measurements. Appl Phys Lett 106:101101
41. Vurgaftman I, Meyer JR (2003) Band parameters for nitrogen-containing semiconductors. J Appl Phys 94:3675–3696
42. Vurgaftman I, Meyer JR, Ram-Mohan LR (2001) Band parameters for III–V compound semiconductors and their alloys. J Appl Phys 89:5815–5875
43. Karpov S (2015) ABC-model for interpretation of internal quantum efficiency and its droop in III-nitride LEDs: a review. Opt Quantum Electron 47:1293–1303
44. Leroux M, Grandjean N, Laugt M et al (1998) Quantum confined Stark effect due to built-in internal polarization fields in (Al, Ga)/N/GaN quantum wells. Phys Rev B 58:R13371–R13374
45. Noh Y-K, Kim M-D, Oh J-E (2011) Reduction of internal polarization fields in InGaN quantum wells by InGaN/GaIn/GaN ultra-thin superlattice barriers with different indium composition. J Appl Phys 110:123108
46. Li Y, Xing Z, Zheng Y et al (2020) High-efficiency near-UV light-emitting diodes on Si substrates with InGaN/GaN/GaIn/GaN multiple quantum wells. J Mater Chem C 8:883–888
47. Cabalu JS, Thomidis C, Friet I, Moustakas TD (2005) Nitride LEDs based on flat and wrinkled quantum wells. In: Razeghi M, Brown GJ (eds) Quantum sensing and nanophotonic devices II, p 185
48. Tsai P-C, Su Y-K, Chen W-R, Huang C-Y (2010) Enhanced luminescence efficiency of InGaN/GaN multiple quantum wells by a strain relief layer and proper Si doping. Jpn J Appl Phys 49:04DG07
49. Zhao Y, Fu H, Wang GT, Nakamura S (2018) Toward ultimate efficiency: progress and prospects on planar and 3D nanostructured nonlinear and semipolar InGaN light-emitting diodes. Adv Opt Photonics 10:246
50. Liang Y-H, Towe E (2018) Progress in efficient doping of high-aluminum-containing group III-nitrides. Appl Phys Rev 5:011107
51. Mnatsakanov TT, Levinstein ME, Pomorskova L et al (2003) Carrier mobility model for GaN. Solid State Electron 47:111–115
52. Hao G-D, Tamari N, Obata T et al (2017) Electrical determination of current injection and internal quantum efficiencies in AlGaN-based deep-ultraviolet light-emitting diodes. Opt Express 25:A639
53. Kim DC, Park JH, Lee JW et al (2015) Overcoming the fundamental light-extraction efficiency limitations of deep ultraviolet light-emitting diodes by utilizing transverse-magnetic-dominant emission. Light Sci Appl 4:e263–e263
54. Zheng J, Liu S, Chou C et al (2015) Direct observation of the biaxial stress effect on efficiency droop in GaN-based light-emitting diode under electrical injection. Sci Rep 5:17227
55. Zheng J, Li J, Zhong Z et al (2017) Effect of electrical injection-induced stress on interband transitions in high Al content AlGaN MQWs. RSC Adv 7:55157–55162
56. Park S-H, Shim J-I (2013) Carrier density dependence of polarization switching characteristics of light emission in deep-ultraviolet AlGaN/AIN quantum well structures. Appl Phys Lett 102:221109
57. Schenk A (1992) A model for the field and temperature dependence of Shockley-Read-Hall lifetimes in silicon. Solid State Electron 35:1563–1596

Publisher’s Note
Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.