Protein-Engineered Polymers Functionalized with Antimicrobial Peptides for the Development of Active Surfaces

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Abstract: Antibacterial resistance is a major worldwide threat due to the increasing number of infections caused by antibiotic-resistant bacteria with medical devices being a major source of these infections. This suggests the need for new antimicrobial biomaterial designs able to withstand the increasing pressure of antimicrobial resistance. Recombinant protein polymers (rPPs) are an emerging class of nature-inspired biopolymers with unique chemical, physical and biological properties. These polymers can be functionalized with antimicrobial molecules utilizing recombinant DNA technology and then produced in microbial cell factories. In this work, we report the functionalization of rPBPs based on elastin and silk-elastin with different antimicrobial peptides (AMPs). These polymers were produced in Escherichia coli, successfully purified by employing non-chromatographic processes, and used for the production of free-standing films. The antimicrobial activity of the materials was evaluated against Gram-positive and Gram-negative bacteria, and results showed that the polymers demonstrated antimicrobial activity, pointing out the potential of these biopolymers for the development of new advanced antimicrobial materials.

Keywords: antimicrobial resistance; antimicrobial surfaces; antimicrobial films; recombinant protein polymers; silk-elastin proteins; elastin-like recombinamers; antimicrobial peptides

1. Introduction

The development of antimicrobial materials is a rapidly growing field, particularly for biomedical applications, demanding new versatile materials. The increasing number of infections caused by resistant bacteria and the low pace of discovery of new and efficient antibiotics necessitates new therapeutic approaches. Moreover, there is an increasing interest in the development of local antimicrobial treatment systems due to improved effectiveness and selectivity.

Natural structural proteins offer a wide range of properties with exceptional physical properties that meet those of synthetic polymers, with additional benefits of enhanced biocompatibility, low immunogenicity and higher degradability [1,2]. The unique physical properties of structural proteins are inherently linked to their composition, relying on the
specific folding of building blocks that hierarchically assemble into well-defined structures. These building blocks have been used as design modules for the formulation of recombinant protein polymers (rPPs) with unique properties [1,3]. Recombinant DNA technology allows the biological synthesis of precisely defined rPPs with absolute control over size and composition. This technology is very powerful, enabling the design of multifunctional protein-based materials by fusing genes encoding for individual protein blocks, and even functional modules, into a single large gene [4]. For instance, the structure of rPPs can be genetically manipulated to incorporate the minimal consensus repeats of silk and elastin into the same molecule [5,6] and to include bioactive domains such as antimicrobial peptides (AMPs) [7–9]. The versatility of this approach allows custom design of the structure of rPPs to obtain functional protein materials that can be processed into different active materials such as coatings, films and fiber mats [7–13].

Due to the unique balance of mechanical properties, biocompatibility and thermostability, elastin-like polypeptides/recombinamers (ELPs/ELRs) and silk-elastin-like proteins/recombinamers (SELPs/SELRs) have been widely studied for the development of biotechnological applications [1,14,15]. ELRs are composed of repeating amino acid building blocks with a typical sequence of VPGXG (V—valine, P—proline, G—glycine, X is any amino acid except for proline) [16], characterized by a reversible temperature-dependent phase-transitional behavior [17,18]. SELPs are a group of copolymers consisting of alternate amino acid repetitions of silk blocks with sequence GAGAGS (A—alanine and S—serine) and elastin blocks, repeated in tandem [6]. These copolymers combine hard (silk) and soft (elastin) domains and can spontaneously form hydrogen-bonded antiparallel β-sheets, with the silk blocks imparting thermal and chemical stability and the elastin blocks reducing the overall crystallinity of the system, increasing its flexibility and water solubility [5,19].

Antimicrobial peptides (AMPs) are promising alternatives to classical antibiotics due to their broad antimicrobial activity and remarkable efficacy [20]. Expressed in almost all living organisms and being part of the innate immune system, AMPs are usually amphipathic short peptides (fewer than 100 amino acid residues), with a positive net charge (+2 to +9) at physiological pH [21,22]. These natural molecules are particularly interesting as therapeutic agents due to their high selectivity toward bacterial cells over mammalian cells [23,24], and low probability to elicit bacterial resistance [25]. Although the underlying mechanisms of action of AMPs are still under active study, it is widely accepted that the basic principle involves the electrostatic interaction of the cationic AMPs with negatively charged components of the bacterial cell membranes [20,26,27]. This electrostatic interaction leads to binding and accumulation of AMPs on the surface of the membranes, promoting membrane depolarization and permeabilization [28]. Several models have been proposed to explain the antimicrobial effect of AMPs and their interaction with the bacterial cell membrane. The most widely described models include the “barrel stave” model, the “toroidal-pore” model and the “carpet” model—several reviews are available on this subject [20–22,28–30].

Due to their broad antimicrobial activity, AMPs have been immobilized onto the surface of materials to confer contact-killing antimicrobial activity [31–34]. Compared with soluble AMPs, the antimicrobial performance of surface-immobilized AMPs is generally reduced due to limited conformational freedom [35,36]. This can be compensated by increasing the density of immobilized AMPs or through introduction of flexible spacers [32,35]. Nevertheless, evidence with AMP-functionalized surfaces (e.g., films and coatings) indicates that the use of AMPs tethered into surfaces [37–41] or using genetically engineered fusion proteins for processing into films or coatings [8,10,12,42] results in surfaces with contact-killing antimicrobial activity by disrupting the integrity of bacteria membranes, resulting in the leakage of intracellular content. The cationic nature of the AMPs probably displaces the divalent cations that maintain the structural integrity of the negatively charged surface of the bacteria, disrupting its integrity; thus, this effect by itself might be sufficient to generate a lethal outcome [43,44].
Our research group is pioneering the formulation of antimicrobial ELR-based materials by genetically engineering an ELR, comprising 200 repetitions of the VPAG sequence (A200), to include the DNA sequence of AMPs [8,12,45]. We have produced antimicrobial ELR-based films through functionalization with the antimicrobial peptides ABP-CM4 [12], a 35 amino acid peptide from the hemolymph of silkworm, and BMAP18 [8], an 18-amino acid truncated variant of bovine myeloid antimicrobial peptide-28. A200 is characterized by an acute thermal hysteresis behavior and the ability to self-assemble into spherical nanoparticles at temperatures above 33 °C [46,47]; we explored this unique behavior for the purification of recombinant hepcidin [45], a 25 amino acid human liver antimicrobial peptide, and suggested its potential application as antimicrobial drug delivery platform.

The aim of this study was to investigate the potential of using both ELRs and SELPs functionalized with AMPs for the development of new antimicrobial surfaces (e.g., films/coatings). Moreover, we also provide a comparison on the antimicrobial performance of SELP- and ELR-based materials obtained while using water and formic acid as solvents for processing. To achieve this, functional SELP hybrids were obtained by using SELP-59-A [6] as a structural module and the antimicrobial peptides ABP-CM4, BMAP18, hepcidin and Synoeca-MP, an antimicrobial peptide isolated from Synoeca surinama social wasp [48], as antimicrobial functional modules. These AMPs were chosen due to their broad and potent antimicrobial activity against different microorganisms [48–54]. While the mechanism of action for Synoeca-MP remains to be elucidated, it has been proposed that ABP-CM4, BMAP18 and hepcidin promote cell membrane depolarization, permeabilization and disruption [51,52,54,55]. SELP-59-A is a protein copolymer designed and synthesized by our research group [6] consisting of nine tandem repetitions of SSE9, where S is the silk block with sequence GAGAGS and E is the elastin block with sequence VPAG. This recombinant protein polymer demonstrated good mechanical properties and cytocompatibility, as well as a high versatility of processing [5,56]. To expand the range of antimicrobial ELR-based materials, we also describe the production of hybrid ELRs comprising A200 as a structural module and hepcidin and Synoeca-MP as functional modules. Hexafluoroisopropanol [57] and formic acid [8,12,13] are commonly used as solvents for rPP processing such as those described here. Here, we explored the use of water as a “greener” and mild alternative for solvent casting. All the recombinant protein materials were used for the production of free-standing films by solvent casting using water and formic acid as solvents and evaluated for their antimicrobial performance against Gram-negative and Gram-positive bacteria following adapted ISO guidelines.

2. Materials and Methods

2.1. Preparation of Genetic Constructions

Genetic constructions were obtained by standard genetic engineering techniques using DNA sequences optimized for Escherichia coli codon usage, following previously described methodologies [45]. Briefly, for A200-based constructions, the genetic sequences coding for AMPs were chemically synthesized with flanking NdeI and KpnI restriction sites (Genscript) and cloned at the N-terminus of A200, previously cloned into a modified pET25b(+) (Novagen) expression plasmid [46]. For the AMP-SELPEL constructions, the genetic sequences for AMPs were chemically synthesized with flanking NdeI restriction sites (Genscript) and cloned at the N-terminus of SELP-59-A, previously cloned into a modified pET25b(+) (Novagen) expression plasmid, containing a poly-histidine tag (6x His) [6]. All the genetic constructions (Tables S1 and S2), Synoeca-A200 (Syn-A200), Synoeca-SELP (Syn-SELP), hepcidin-A200 (Hep-A200), hepcidin-SELP (Hep-SELP), ABP-CM4-SELP (CM4-SELP) and BMAP18-SELP were confirmed by DNA sequencing (Eurofins). The final expression plasmids were transformed into E. coli BL21(DE3) for recombinant protein production.
2.2. Protein Production and Purification

All proteins—Hep-A200, MW = 87.9 kDa; Syn-A200, MW = 86.8 kDa; Hep-SELP, MW = 60.0 kDa; Syn-SELP, MW = 58.6 kDa; CM4-SELP, MW = 60.8 kDa; BMAP18-SELP, MW = 59.2 kDa)—were expressed in *E. coli* BL21(DE3) by means of auto-induction [6]. Cell cultures were allowed to grow for 22 h at 37 °C and 200 rpm, with a volumetric ratio of 1:4 in Terrific Broth (12 g tryptone, 24 g yeast extract, 5.04 g glycerol, 12.54 g K$_2$HPO$_4$, 2.31 g KH$_2$PO$_4$, per liter), supplemented with 2 g/L lactose for auto-induction (TBlac), and 100 mg/L ampicillin (AMP-SELP) or 50 mg/L kanamycin (AMP-A200).

After protein production, bacterial cells were collected by centrifugation, resuspended in TE buffer solution (50 mM Tris-HCl + 1 mM EDTA at pH 8.0) and disrupted by sonication using a Vibra cell™ 75043 ultrasonic liquid processor (Bioblock Scientific) with a 25 mm diameter probe (3 s pulse on, 9 s pulse off, total sonication time: 10 min). The pH of the crude cell lysate was adjusted to 3.5 with 1.6 M HCl to precipitate endogenous *E. coli* contaminants [6,45] and removed by centrifugation at 11,500 × g for 20 min at 4 °C. Samples were kept at ice-cold temperature during the entire process. Purification of AMP-A200 proteins (Hep-A200 and Syn-A200) was achieved by means of inverse transition cycling (ITC) based on previously described methodologies [45,46]. Briefly, the AMP-A200 proteins were subjected to three cycles of hot (37 °C) and cold (4 °C) incubation (2 h at each temperature) and centrifugation steps (11,500 × g for 20 min) in ultrapure water (mQ; Milli-Q®, Millipore). AMP-SELP proteins were purified by ammonium sulfate precipitation with 20% saturation followed by dialysis against mQ water (refer to [6] for detailed description), using natural cellulose with a 12,000–14,000 Da molecular weight cut-off (Medicell Membranes Ltd.). Protein production and purification was assessed by sodium dodecyl sulfate-polyacrylamide gel electrophoresis (SDS-PAGE) on a 10% acrylamide gel with 0.3 M copper chloride (CuCl$_2$) staining. Purified polymer fractions were freeze-dried (Christ Alpha 2-4 LD Plus from Bioblock Scientific) and stored at room temperature prior to use.

2.3. Western Blot Analysis of AMP-SELP

Following 10% SDS-PAGE electrophoresis separation of pure solutions of AMP-SELP, protein bands were transferred onto a nitrocellulose membrane at 100 V for 1 h. The membrane was then blocked for 1 h with 5% non-fat dried milk in TBS-T buffer (50 mM Tris-HCl, 150 mM NaCl, 0.05% Tween 20 at pH 8), washed with TBS-T and incubated overnight with mouse monoclonal anti-polyHistidine antibody (Sigma-Aldrich) diluted 1:3000 in the blocking solution. Following incubation with the primary antibody, membrane was washed with TBS-T and incubated with goat anti-mouse IgG peroxidase (HRP) antibody (Sigma-Aldrich) diluted 1:5000 in the blocking solution. Protein detection was assessed by chemiluminescence using the ECL detection system (Merck Millipore) and a Chemi-Doc XRS system (BioRad).

2.4. Preparation of Free-Standing Films

Films of AMP-A200 and AMP-SELP were prepared by solvent casting using double-distilled water (ddH$_2$O) or formic acid (98%–100%, Merck) as solvents. Pure lyophilized protein was dissolved in the corresponding solvent at ice-cold temperature to concentrations of 10% (w/v) for AMP-A200 and 3% (w/v) for AMP-SELP, and mixed until complete dissolution was achieved. A volume of 100 µL of each protein solution was cast onto polytetrafluoroethylene (PTFE, Teflon®) molds with 10 mm diameter and allowed to dry at room temperature under extraction. After complete solvent evaporation, films were gently peeled off from the molds. The AMP-SELP free-standing films were subjected to a post-processing treatment with methanol-saturated air at room temperature for 48 h in a desiccator to promote water insolubility [5] and air-dried at room temperature for at least 48 h before characterization.
2.5. Structural Characterization of AMP-SELP Films

Structural changes induced by the methanol treatment in the AMP-SELP films were assessed by attenuated total reflection Fourier-transform infrared (ATR-FTIR) spectroscopy using a Perkin Elmer Spectrum Two™ spectrometer with a DTGS (deuterated triglycine sulfate) detector, coupled with a single reflection diamond UATR (universal attenuated total reflectance, Perkin Elmer). Spectra were collected in the range between 4000 and 600 cm\(^{-1}\), after 64 scans with a resolution of 4 cm\(^{-1}\) and automatic correction of atmospheric CO\(_2\)/H\(_2\)O. Spectra analysis was conducted in the amide I band region (1600–1700 cm\(^{-1}\)) by second derivatization and Gaussian curve fitting using OriginPro 9.0 software (OriginLab, Northampton, MA) [58]. For a reliable comparison between the secondary structure assignment of the different samples, curve fitting was performed with the same set of parameters. The contribution of each fitted component to the amide I band was estimated by integration of the area under the curve and normalizing for the total area of amide I.

2.6. Evaluation of Antimicrobial Activity

The antimicrobial activity of the polymeric films was assessed by direct contact against *E. coli* HB101 and *S. aureus* ATCC6538 using an adapted version of ISO 22196 [12]. Films were sterilized by UV exposure for 30 min and transferred to sterile non-treated polystyrene (PS) flat bottom 24 well plates. The surface of the films was inoculated with 50 μL of bacterial cell suspension (1 × 10\(^6\) CFUs/mL) and incubated for 120 min at 37 °C. After incubation, 950 μL of sterile 0.87% NaCl solution was added and carefully agitated by gently pipetting up and down 10 times, followed by plating 100 μL of serial diluted cell suspensions on LB agar (10 g tryptone, 5 g yeast extract, 10 g NaCl, 15 g agar, per liter). Cell suspensions inoculated in empty wells (without protein films) were used as negative control for antibacterial activity. Antimicrobial assays were performed in triplicate at different days with three replicates each. Results were expressed as lethality percentage (% kill) calculated by the equation:

\[
\% \text{ kill} = \frac{\text{CFUs (control)} - \text{CFUs (sample)}}{\text{CFUs (control)}} \times 100
\]

2.7. Statistical Analysis

Two-way analysis of variance (ANOVA) with Bonferroni’s post-test and 95% confidence level was carried out with GraphPad Prism 6 software. Results were presented as mean values ± standard deviation (SD), and \( p \leq 0.05 \) was considered statistically significant. All experiments were performed in triplicate and repeated at least three times in independent days.

3. Results and Discussion

3.1. Production and Purification of Recombinant AMP-A200 and AMP-SELP

All recombinant proteins (Hep-A200, Syn-A200, Hep-SELP, Syn-SELP, CM4-SELP and BMAP18-SELP) were successfully produced in *E. coli* BL21(DE3) at high expression levels (see Figures 1b and 2a for representative samples).
Figure 1. Purification of AMP-A200-based protein polymers. (a) Schematic representation of the purification process by ITC. (b) Representative SDS-PAGE analysis of purification for Syn-A200. After sonication and acidification to pH 3.5, the insoluble debris was removed by centrifugation and the recombinant protein polymer was purified from the clear supernatant (lane 1) by ITC using hot/cold cycles. For each cycle, the protein-enriched solution was heated to 37 °C and then centrifuged at the same temperature. The resulting hot supernatant was discarded (lanes 2, 4 and 6) and the precipitated protein was resuspended in ice-cold ddH₂O. After complete resuspension, the solution was centrifuged at 4 °C, and the resulting cold supernatant (lanes 3, 5 and 7) was submitted to a new cycle. A pure protein polymer fraction (indicated by the white arrow) was obtained after three complete cycles. MWM—molecular weight marker (Precision Plus Protein Unstained Standards from Bio-Rad) in kDa; 1—cycle one, cold supernatant; 2—cycle one, hot supernatant; 3—cycle two, cold supernatant; 4—cycle two, hot supernatant; 5—cycle three, cold supernatant; 6—cycle three, hot supernatant. Gel was stained with 0.3 M CuCl₂.

Figure 2. (a) Representative SDS-PAGE analysis of the purification process for BMAP18-SELP by salting out with ammonium sulfate at 20% saturation: MWM—molecular weight marker (GRS Unstained Protein Marker from Grisp) in kDa; 1—crude cell lysate after sonication; 2—supernatant after acidification at pH 3.5; 3—supernatant after salting out; 4—fraction of the precipitated sample after salting out; 5—dialyzed protein sample. Gel was stained with 0.3 M CuCl₂. (b) Representative SDS-PAGE (image on the left) and Western blot (image on the right) analysis of pure BMAP18-SELP (6) with unstained SDS-PAGE standard molecular weight marker (MWM, in kDa; BioRad).

Purification of the A200-based protein polymers was achieved by ITC, resulting in highly pure protein factions (Figure 1). After three heating/cooling cycles (Figure 1a), a pure protein fraction was obtained, as demonstrated by presence of a single band around 100 kDa (lane 7, Figure 1b).
Purification of AMP-SELP was achieved by a non-chromatographic method involving the use of ammonium sulfate at 20% saturation for the selective precipitation of SELP [6]. Analysis of the electrophoretic pattern of purified BMAP18-SELP (used in Figure 2 as representative sample) revealed the presence of two strong bands at approximately 60 and 80 kDa and some less intense bands at lower molecular weights (Figure 2a). Western blot analysis using antibodies to the His tag revealed a similar band profile (Figure 2b), characterized by a strong band corresponding to BMAP18-SELP and unknown lower molecular weight protein bands. Overall, ammonium sulfate precipitation demonstrably resulted in pure AMP-SELP fractions.

The same electrophoretic pattern and Western blot results were observed for the remaining SELP-based constructions (Figure S1). The abnormal gel mobility to higher molecular weights than expected is a consequence of the hydrophobic nature of the proteins and has been previously observed for other recombinant protein polymers [6,45,46,59,60]. After purification and lyophilization, the lyophilized product was weighed, resulting in volumetric productivities per liter of production of 50 mg/L for Hep-A200, 90 mg/L for Syn-A200, 90 mg/L for Hep-SELP, 73 mg/L for Syn-SELP, 120 mg/L for CM4-SELP and 70 mg/L for BMAP18-SELP.

3.2. Secondary Structure Analysis of AMP-SELP Films

Maintenance of structural integrity is fundamental in the development of materials, especially for applications that require contact with aqueous environments such as culture media or body fluids. Due to the intrinsic thermal hysteresis behavior of A200, both Hep-A200 and Syn-A200 films were stable over a wide range of temperatures without the need of crosslinking agents to promote structure stabilization [12,13]. However, as-cast SELP materials based on the SELP-59-A backbone are highly water soluble, limiting their potential application. Nonetheless, this limitation can be surpassed by treatment with methanol, leading to water insolubility by inducing a physical crosslinking through the formation of hydrogen-bonded β-sheets [5,56]. In this study, the AMP-SELP films were exposed to a methanol-saturated atmosphere for 48 h, and structural changes were evaluated by ATR-FTIR spectroscopy. Secondary structure analysis was performed based on the amide I region (1600 to 1700 cm⁻¹), a sensitive spectral region used for protein studies, which occurs mostly from C=O stretching vibrations [61–63]. For all samples, the infrared spectrum of the non-treated films was characterized by a broad amide I absorption band centered at approximately 1628 cm⁻¹ with a pronounced shoulder around 1646 cm⁻¹ (Figure S2), indicating the presence of β-sheets and a relevant contribution from random coils, respectively [5]. After the methanol treatment, the amide I band was narrower and shifted to approximately 1623 cm⁻¹, indicating that major contributions arise from β-structures and less from other secondary structures [5,58]. Since the broad amide I band collected from spectra collected results from the overlapping components arising from the various secondary structure elements [64], band-narrowing techniques such as second derivative analysis are pivotal for a more detailed characterization of protein conformation, thus providing the basis for the quantitative estimation of protein secondary structure [65–67]. Quantitative secondary structure analysis of AMP-SELPs was performed by second derivative analysis and curve fitting methods, with assignment of structural conformations by reference to literature [58,63] (Table S3).

After the methanol treatment, the β-sheet content of all films demonstrated an overall increase of more than 8% for films prepared with water and more than 3% for films prepared with formic acid in relation to the untreated samples (Table 1). These results indicate that methanol induced a molecular reorganization (Figure S2), leading to the formation of additional hydrogen-bonded β-sheets that resulted in water insolubility.
Table 1. Beta-structure content of AMP-SELP films, produced from water and formic acid solutions, after and before methanol (MetOH) treatment. Results presented as fractional distribution of total β-structure content determined by second derivative and Gaussian curve fitting.

| Sample         | Water Untreated | MetOH-Treated | Formic Acid Untreated | MetOH-Treated |
|----------------|-----------------|---------------|-----------------------|---------------|
| Hep-SELP       | 45.62%          | 54.91%        | 44.90%                | 48.48%        |
| Syn-SELP       | 38.40%          | 54.56%        | 48.70%                | 51.86%        |
| CM4-SELP       | 40.86%          | 49.75%        | 44.42%                | 55.68%        |
| BMAP18-SELP    | 37.62%          | 55.02%        | 45.95%                | 52.7%         |

3.3. Antimicrobial Activity of AMP-ELR and AMP-SELP Films

The antimicrobial activity of Hep-A200, Syn-A200, Hep-SELP, Syn-SELP, CM4-SELP and BMAP18-SELP films (produced from aqueous or formic acid solutions) was evaluated against *S. aureus* ATCC6538 (Gram-positive) and *E. coli* HB101 (Gram-negative) by direct contact after 120 min.

Analysis of antibacterial assays reveals that antimicrobial performance is highly dependent on the structural (ELR or SELP) and functional (AMP) modules and less dependent on the solvent used for film production, although films produced with formic acid display slightly higher average % kill values. Overall, the AMP-A200 films demonstrated better antimicrobial performance than the AMP-SELP films, independently of the solvent used (Figure 3).

Figure 3. Antimicrobial activity of AMP-SELP and AMP-A200 hybrid materials assessed by direct contact with the surface of the films against (a) *Escherichia coli* and (b) *Staphylococcus aureus*. Cell suspensions inoculated in empty wells were used as reference for 100% survival. Error bars represent means ± SD (ns, non-significant; * p ≤ 0.05; ** p ≤ 0.01; *** p ≤ 0.001; **** p ≤ 0.0001). Experiments were performed in triplicate in independent days with three replicates each (see Tables S4 and S5 for CFU enumeration). No statistical significance was found between films prepared using water or formic acid as solvents.

Syn-SELP and Hep-SELP films produced using formic acid as solvent show average lethality percentage (% kill) values for *E. coli* near 17%, whereas Syn-A200 and Hep-A200 films produced using the same solvent demonstrated average values of approximately 96% and 88%, respectively (Figure 3a, Table 2). This same trend is observed for CM4-SELP and BMAP18-SELP compared to reference values obtained in previous works [8,12] (Table 2). The SELP-based films produced from formic acid present values of % kill around 35% and 60% for CM4-SELP and BMAP18-SELP, respectively, whereas this value increases to 86% and 100% using A200 as a structural backbone (Table 2). The distinct antimicrobial performance observed for AMP-A200 and AMP-SELP films is possibly due to the different conformations adopted by the structural modules (SELP and ELR) in the protein polymers. ELRs adopt a highly hydrated structure with conformational
flexibility [68,69]. This flexibility favors the ability of the AMPs to interact with bacteria. In contrast, the methanol-treated SELPs adopt a more rigid crystalline structure, characterized by antiparallel hydrogen-bonded \( \beta \)-sheet structures (silk II-like) [5,70,71]. As the AMPs are fused to the N-terminal of the SELP, that is, to the silk blocks, the conformational flexibility of the AMPs is therefore restricted, limiting the ability of AMP moieties to contact with bacteria.

Table 2. Percentage lethality (% kill) of the different protein polymer films after 2 h incubation at 37 \( ^\circ \)C with *Escherichia coli* and *Staphylococcus aureus*. Films were produced by solvent casting using (a) water and (b) formic acid as solvents. Experiments were performed in triplicate in independent days with three replicates each.

| Sample          | % Kill (*E. coli*) | % Kill (*S. aureus*) | Reference |
|-----------------|--------------------|----------------------|-----------|
| Hep-A200 (a)    | 69.9 ± 10.2        | 60.8 ± 23.5          | This work |
| Hep-A200 (b)    | 87.7 ± 8.3         | 76.7 ± 13.7          | This work |
| Syn-A200 (a)    | 85.9 ± 9.0         | 75.4 ± 25.4          | This work |
| Syn-A200 (b)    | 95.9 ± 6.2         | 85.6 ± 8.2           | This work |
| CM4-A200 (b)    | 86.5 ± 0.7         | 69.7 ± 4.5           | [12]      |
| BMAP18-A200 (b) | 100.0              | 98.8 ± 2.0           | [8]       |
| Hep-SELP (a)    | 11.4 ± 5.5         | 33.9 ± 41.9          | This work |
| Hep-SELP (b)    | 17.7 ± 6.1         | 65.9 ± 30.8          | This work |
| Syn-SELP (a)    | 28.9 ± 4.9         | 29.3 ± 44.7          | This work |
| Syn-SELP (b)    | 16.9 ± 5.8         | 30.3 ± 62.2          | This work |
| CM4-SELP (a)    | 28.1 ± 6.8         | 10.7 ± 59.6          | This work |
| CM4-SELP (b)    | 34.8 ± 8.9         | 30.1 ± 4.8           | This work |
| BMAP18-SELP (a) | 73.7 ± 4.2         | 41.5 ± 35.3          | This work |
| BMAP18-SELP (b) | 60.4 ± 14.4        | 48.4 ± 24.5          | This work |

Comparing the mean values of lethality percentage (% kill) against the tested bacteria, best results were generally obtained against *E. coli* than against *S. aureus* (Figure 3). The antibacterial assays against *S. aureus* demonstrated high standard deviation (SD) values, which are a consequence of the ability of *S. aureus* to grow as planktonic aggregates [72], thus reducing the direct contact with the films. This observation has been previously reported in a similar direct contact assay using CM4-A200 films [12] and fiber mats [13]. Indeed, scanning electron microscopy micrographs clearly demonstrated the ability of *S. aureus* to form aggregates reducing the contact of the overall microbial community with the film surface [12]. Similarly, in the present study, the antimicrobial action of the films was mediated by direct contact, in which cells must be available to contact with the film surface. If cells are able to form aggregates, only those in contact with the film surface are subjected to an antimicrobial effect, explaining the high variability of the assay. Still, analysis of the mean values demonstrates that, overall, the AMP-ELR films demonstrate a better antibacterial performance than AMP-SEL films.

In addition to the structural backbone (ERL or SELP), the antimicrobial activity was also influenced by the type of AMP. Comparing the effect of the functional module in the antibacterial activity of the AMP-SEL films against *E. coli*, best results were obtained with BMAP18 with average % kill values greater than 60%, whereas the lowest antimicrobial performance was observed for Hep-SELP with % kill values lower than 20% (Table 2). The apparent low antimicrobial performance of Hep-SELP films may be due to the secondary structure of hepcidin. Unlike Synoeca, CM4 and BMAP18, which show a linear \( \alpha \)-helix secondary structure in membrane-mimicking environments [48,73,74], hepcidin forms a less flexible \( \beta \)-hairpin structure, with a central core cross-linked by four disulfide bonds [75] that might intertwine with SELP \( \beta \)-sheets, compromising the peptide interaction with bacteria membranes.

Regarding the assays performed with *E. coli* (Figure 3a), results revealed that, for the confidence interval considered, all polymer films (except for Hep-SELP films prepared
using water as solvent) are statistically significant, demonstrating the antimicrobial effect of these films.

The antimicrobial activity observed for the fusion proteins is attributed to the functional module (AMPs) and not to the structural module (A200 or SELP). Previously, we have demonstrated that CM4-A200 displays potent antimicrobial activity against *P. aeruginosa*, reaching lethality percentage values of ca. 97% [12]. When compared with other A200-based fusion proteins (BMP2-A200: bonemorphogenetic protein-2 linked with A200 and prosubtilisin-A200: prosubtilisin enzyme linked with A200) the antimicrobial activity was found to be very low (ca. 20% for BMP2-A200) or even absent (proliferation of ca. 14% for prosubtilisin-A200) [12]. These results indicated that the antimicrobial performance was attributed to the AMP and not to the structural A200. In a similar way, the antimicrobial activity of non-functionalized SELPs (SELP without AMPs) demonstrated to be statistically non-significant compared with the control (Figure S3). Finally, the attribution of antimicrobial activity to the diffusion of eroded molecules is very unlikely as antimicrobial assays utilizing the Kirby–Bauer test demonstrated the absence of inhibition halos (unpublished data). We have previously demonstrated that CM4-A200 and BMAP18-A200 do not show inhibition halos in disk diffusion assays (Kirby–Bauer test) but exert a strong antimicrobial effect mediated by direct contact [8,12].

4. Conclusions

In this study, we report the fabrication of new antimicrobial protein-based polymers with different structural backbones (structural module), namely SELP-59-A, a silk-elastin-like protein able to form self-structured β-sheet structures, and A200, an elastin-like recombinamer characterized by a more flexible conformation. The protein polymers were genetically engineered to include functional modules consisting of the amino acid sequences for antimicrobial peptides hepcidin, ABP-CM4 and BMAP18, as well as the recently isolated Synoeca-MP (Table S6). All constructs were successfully produced in *E. coli*, purified by simple non-chromatographic approaches and processed into free-standing films using water and formic acid as solvents. No statistically significant differences were observed in the antimicrobial activity of films produced with water or with formic acid, indicating the feasibility of using a water as mild and “green” alternative to more harsh chemicals. However, evaluation of the antimicrobial activity of the films indicates that the structural backbone plays a central role in antimicrobial performance suggesting that AMPs should be conformationally free to exert their antimicrobial effect. While the AMP-ELR films were highly effective in exerting an antimicrobial effect against *E. coli* and *S. aureus*, we propose that the more rigid structure of the AMP-SELP materials restricts the ability of the AMPs to interact with bacterial cells and consequently results in a more limited performance. A strategy to overcome this limitation could be through the inclusion of flexible glycine/serine spacers in the genetic design to improve the conformational freedom of the AMPs [76,77], thereby increasing their accessibility. Bagheri et al. [36] demonstrated that the presence of flexible spacers and their length are determinant factors for antimicrobial activity since the presence of the spacer can enable peptide insertion into the cell. On the other hand, the antimicrobial activity of AMPs can be lost when immobilized on surfaces without the presence of a spacer [38,78]. This supports our assumption that increased antibacterial activity can be achieved by introducing flexible spacers in the genetic design. Nevertheless, SELP-based antimicrobial biomaterials can be promising opportunity for novel materials, as SELPs combine the elasticity, resilience and solubility of elastin with the mechanical strength and versatility of processing of silk [5,79].

Analysis of antimicrobial results against *S. aureus* revealed a high variability because of its ability to form cell aggregates thus reducing the contact with the antimicrobial surface. This high variability could be reduced by lowering the number of cells in the inoculum, so as to ensure that the majority of bacteria are placed in contact with the film surface. As such, although the antimicrobial assays were performed based on ISO 22196, we suggest
that results should be critically analyzed upon assessing the antimicrobial performance of surfaces while using cells with the ability to grow in aggregates.

Overall, the findings of the present study represent an important contribution towards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10.3390/app11125352/s1, Figure S1: SDS-PAGE (a) and Western blot (b) analysis of pure BMAP18-wards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials. Supplementary Materials: The following are available online at https://www.mdpi.com/article/10.3390/app11125352/s1, Figure S1: SDS-PAGE (a) and Western blot (b) analysis of pure BMAP18-wards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10.3390/app11125352/s1, Figure S1: SDS-PAGE (a) and Western blot (b) analysis of pure BMAP18-wards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10.3390/app11125352/s1, Figure S1: SDS-PAGE (a) and Western blot (b) analysis of pure BMAP18-wards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10.3390/app11125352/s1, Figure S1: SDS-PAGE (a) and Western blot (b) analysis of pure BMAP18-wards the rational design of genetically engineered protein polymers functionalized with antimicrobial peptides and provide cues for a next generation of antimicrobial materials.
Informed Consent Statement: Not applicable.

Data Availability Statement: Data is contained within the article or supplementary information.

Conflicts of Interest: The authors declare no conflict of interest.

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