Near Infrared-Emitting Cr\(^{3+} \)/Eu\(^{3+}\) Co-doped Zinc Gallogermanate Persistence Luminescent Nanoparticles for Cell Imaging

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**Abstract**

Near infrared (NIR)-emitting persistent luminescent nanoparticles have been developed as potential agents for bioimaging. However, synthesizing uniform nanoparticles with long afterglow for long-term imaging is lacking. Here, we demonstrated the synthesis of spinel structured Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\):Cr\(^{3+}\) (ZGGO:Cr\(^{3+}\)) and Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\):Cr\(^{3+}\),Eu\(^{3+}\) (ZGGO:Cr\(^{3+}\),Eu\(^{3+}\)) nanoparticles by a sol-gel method in combination with a subsequent reducing atmosphere-free calcination. The samples were investigated via detailed characterizations by combined techniques of XRD, TEM, STEM, selected area electron diffraction, photoluminescence excitation (PLE)/photoluminescence (PL) spectroscopy, and temperature-dependent PL analysis. The single-crystalline nanoparticles are homogeneous solid solution, possessing uniform cubic shape and lateral size of ~80–100 nm. Upon UV excitation at 273 nm, ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) exhibited a NIR emission band at 697 nm (\(^{2}E\) \(\rightarrow\) \(^{4}A_2\) transition of distorted Cr\(^{3+}\) ions in gallogermanate), in the absence of Eu\(^{3+}\) emission. NIR persistent luminescence of the sample can last longer than 7200 s and still hold intense intensity. Eu\(^{3+}\) incorporation increased the persistent luminescence intensity and the afterglow time of ZGGO:Cr\(^{3+}\), but it did not significantly affect the thermal stability. The obtained ZGGO:Cr\(^{3+}\),Eu\(^{3+}\)-NH\(_2\) nanoparticles possessed an excellent imaging capacity for cells in vitro.

**Keywords:** Near infrared (NIR)-emitting, Persistent luminescent, Bioimaging, Spinel structure, Nanoparticles

**Background**

Persistent luminescent materials can emit for a long time up to hours after the stoppage of excitation [1]. Mainly due to their great research interests, the phosphors have been commercialized as night or dark environment vision materials for a wide range of applications such as security signs, emergency route signage, identification markers, or medical diagnostics [2]. The typical long-persistent phosphors are the commercialized primary color emitters, such as the red Y\(_2\)O\(_2\)S:Eu\(^{3+}\),Mg\(^{2+}\),Ti\(^{4+}\), or CaS:Eu\(^{2+}\),Tm\(^{3+}\),Ce\(^{3+}\) [3, 4], the green SrAl\(_2\)O\(_4\):Eu\(^{2+}\),Dy\(^{3+}\), or MgAl\(_2\)O\(_4\):Mn\(^{2+}\) [5, 6], and the blue CaAl\(_2\)O\(_4\):Eu\(^{2+}\),Nd\(^{3+}\), or SrMgSi\(_2\)O\(_6\):Eu\(^{3+}\),Dy\(^{3+}\) [7, 8] phosphors. Although many successes have been made in visible persistent phosphors, the investigation and development of persistent phosphors in the near infrared (NIR) region (~700–2500 nm) are insufficient. In recent years, the potential applications of persistent phosphors showing red or NIR luminescence have expanded from night-vision security signs to in vivo imaging systems [1, 9, 10]. Persistent luminescent materials with attached photosensitizers as an in vivo agent were firstly tried by Chen and Zhang for photodynamic therapy [11]. Then, Scherman et al. reported a milestone work on in vivo bioimaging with the NIR-emitting phosphor of Ca\(_{0.2}\)Zn\(_{0.9}\)Mg\(_{0.9}\)Si\(_2\)O\(_6\):Eu\(^{2+}\),Mn\(^{2+}\),Dy\(^{3+}\) [12]. Soon afterwards, two new NIR-emitting phosphors of CaMgSi\(_2\)O\(_6\):Eu\(^{2+}\),Mn\(^{2+}\),Pr\(^{3+}\) and Ca\(_2\)Si\(_8\)N\(_8\):Eu\(^{2+}\),Tm\(^{3+}\) with improved performance have been developed by the same group [13, 14]. Recently, Cr\(^{3+}\)-doped gallate persistent phosphors with NIR emission and long afterglow, including spinel ZnGa\(_2\)O\(_4\):Cr\(^{3+}\) and their variants, such as Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\):Cr\(^{3+}\), Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\):Cr\(^{3+}\),Yb\(^{3+}\),Er\(^{3+}\), and ZnGa\(_2\)-x(Ge/Sn)yO\(_4\):Cr\(^{3+}\), were prepared by a solid-state method [1, 9, 10, 15–21]. The ceramic-disc samples exhibited afterglow time up to 360 h at the NIR region.
but the bulky materials are unsuitable for in vivo bioimaging. NIR-emitting long-persistent luminescent nanoparticles of ZnGa$_2$O$_4$:Cr$^{3+}$ [22, 23], ZnGa$_2$O$_4$:Cr$^{3+}$,Sn$^{4+}$ [19–21], and Zn$_{2.94}$Ga$_{1.96}$Ge$_2$O$_{10}$:Cr$^{3+}$,Pr$^{3+}$ [9] were synthesized by a sol-gel method in combination with a subsequent reducing atmosphere-free calcination. The persistent luminescence of the nanoparticles powder exhibits bright NIR luminescence in the biological transparency window with a superlong afterglow time. PEGylation greatly improves the biocompatibility and water solubility of the nanoparticles, which hold great potential for long-term in vivo bioimaging application with high SNR without the need for in situ excitation. It is believed that ions selected from a group consisting of alkaline earth ions, lanthanide ions, and Li$^+$ co-doping with Cr$^{3+}$ in zinc gallate and zinc gallogermanate would yield remarkable NIR persistent luminescence [1]. Eu$^{3+}$ in oxide hosts always exhibits a red emission at ~700 nm arising from the $^5D_0$-$^7F_4$ intra-4f electronic transition upon short UV excitation into the charge transfer (CT)
band at 250 nm [24]. On the other hand, Cr$^{3+}$ is a favorable luminescent center in solids because of its narrowband emissions (usually at 700 nm) due to the spin-forbidden $^2E-^4A_2$ transition, or a broadband emission (650–1600 nm) due to the spin-allowed $^4T_2-^4A_2$ transition [1, 20]. In view of these, Cr$^{3+}$/Eu$^{3+}$ co-doped zinc gallate and zinc gallogermanate would yield intense NIR persistent luminescence, owing to that the charge transfer band (CTB) of O$_2^-$-Eu$^{3+}$ overlaps with the CTB of O$_2^-$-Ga$^{3+}$, and the emission at ~700 nm from $^5D_0-^7F_4$ transition of Eu$^{3+}$ overlaps with that from $^2E-^4A_2$ transition of Cr$^{3+}$. Furthermore, Eu$^{3+}$ ions substituting for Ga$^{3+}$ ions in distorted octahedral sites may yield suitable host crystal-field strength around Cr$^{3+}$ ions, thus affecting the NIR emission. In this work, Zn$_3$Ga$_2$Ge$_2$O$_{10}$:Cr$^{3+}$,Eu$^{3+}$ (termed as ZGGO:Cr$^{3+}$,Eu$^{3+}$) nanoparticles were synthesized by a sol-gel method in combination with a subsequent reducing atmosphere-free calcination, which would be used as the promising nanoprobes for future bioimaging. The samples were investigated via detailed characterizations by combined techniques of X-ray diffractometry (XRD), transmission electron microscopy (TEM), STEM, selected area electron diffraction (SAED), photoluminescence excitation (PLE)/photoluminescence (PL) spectroscopy, and temperature-dependent PL analysis. In the following sections, we report the synthesis, characterization, and application of the ZGGO:Cr$^{3+}$,Eu$^{3+}$ nanoparticles.

**Experimental**

**Synthesis**

The starting metal sources are Zn(NO$_3$)$_2$·6H$_2$O, Cr(NO$_3$)$_3$·9H$_2$O, Ga$_2$O$_3$, Eu$_2$O$_3$, and GeO$_2$ were all 99.99% pure products purchased from Sinopharm.
(Shanghai, China). The other reagents are of analytical grade and were purchased from Shenyang Chemical Reagent Factory (Shenyang, China). Zn(NO$_3$)$_2$·6H$_2$O and Cr(NO$_3$)$_3$·9H$_2$O were dissolved in deionized water. Ga$_2$O$_3$ and Eu$_2$O$_3$ were dissolved in nitric acid solution. GeO$_2$ and ethylenediaminetetraacetic acid (EDTA) were dissolved in dilute ammonium hydroxide. To the mixture solution, EDTA solution was slowly added without any precipitation, and the molar ratio of total metal ions to EDTA was maintained at 1:2. The atom molar ratio of Zn:Ga:Ge:Cr:Eu was fixed to be 3:1.984:2:0.01:0.006. The final solution was vigorously stirred at room temperature for 1 h, then heated in an oven at 85 °C for the slow evaporation of water until the solution became a sol that finally became a gel. The obtained gel was heated at 200 °C for 3 h to form black porous materials. Finally, the porous materials were ground and annealed under flowing O$_2$ gas (200 mL/min) at selected temperatures for 2 h.

**Surface Functionalization**

The ZGGO:Cr$^{3+}$,Eu$^{3+}$ powder was ground for 30 min, then 150 mg of the obtained sample was added into 50 mL of 0.1 mol/L NaOH solution. After sonication for 1 h, the suspension was vigorously stirred for 24 h at room temperature. The resulting colloid solution was centrifuged at 1000 rpm for 10 min to remove large size particles, and the supernatant was centrifuged at 10,000 rpm for 10 min to collect the precipitate. The as-obtained ZGGO:Cr$^{3+}$,Eu$^{3+}$-OH nanoparticles were washed with deionized water three times.

Ten milligrams of ZGGO:Cr$^{3+}$,Eu$^{3+}$-OH nanoparticles was dispersed in 4 mL of dimethylformamide (DMF) with the assistance of sonication for 10 min. Then, 40 μL of 3-aminopropyl-triethoxysilane (APTES) was added under vigorous stirring for 24 h at room temperature. The as-obtained ZGGO:Cr$^{3+}$,Eu$^{3+}$-NH$_2$ nanoparticles were collected by centrifugation at 10,000 rpm for 10 min and washed with DMF three times to remove unreacted APTES.

**Cell Imaging**

Hek293T cells were cultured in DMEM with 10% FBS and seeded in 35-mm culture dishes for 2 h in a CO$_2$ incubator. The as-obtained ZGGO:Cr$^{3+}$,Eu$^{3+}$-NH$_2$ nanoparticles were dispersed in cell medium (50 mg/mL), which were excited for 10 min by a 254-nm UV lamp and then moved to culture dishes treated for 1 h. After removing the cell medium, 0.1 mL of 1% formaldehyde-PBS was added and the cells were stained with 0.5 mL DAPI dye in the dark for 10 min. Finally, the cells were washed with PBS several times for further characterization.

All studies involving animals were approved by the university animal care and use committee.

**Characterization Techniques**

Phase identification was performed by XRD (model SmartLab; Rigaku, Tokyo, Japan) operating at 40 kV/40 mA using nickel-filtered Cu Kα radiation and a scanning speed of 6°/min. Morphologies of the products were observed via TEM (model JEM-2000FX; JEOL, Tokyo). Photoluminescence of the phosphors was analyzed with an FP-8600 fluorospectrophotometer (JASCO, Tokyo). The persistent luminescence signals were obtained using Horiba JY FL3-21. The afterglow decay images were recorded in a dark room using a Kodak In-Vivo Imaging System FX Pro. The cell imaging was conducted by a laser scanning confocal microscope (LEICA TCS SP2, Germany).

All studies involving animals were approved by the university animal care and use committee.
Results and Discussion

The phase purity of the samples was first investigated by XRD. Figure 1 (up) shows the XRD patterns of the as-prepared ZGGO:Cr\(^{3+}\) and ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) calcined at 1000 °C, which were identified along with the spinel structured Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\) [1, 9]. The crystal structure of Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\) is the same with that of ZnGa\(_2\)O\(_4\) (JCPDS No. 38-1240), which is the solid solution of ZnGa\(_2\)O\(_4\) and Zn\(_2\)GeO\(_4\). In the structure of Zn\(_3\)Ga\(_2\)-Ge\(_2\)O\(_{10}\), Ge plays the role of substitution of Ga, conducive to the formation of traps, while ZnGa\(_2\)O\(_4\) is the dominant crystal structure [1]. There are two kinds of cations in one-unit cell; Zn\(^{2+}\) and Ga\(^{3+}\) are surrounded by four and six oxygen anions forming a tetrahedron and an octahedron, respectively (Fig. 1, below). Calculations from the diffraction data yielding the cell constants for ZGGO:Cr\(^{3+}\) are \(a = b = \sim 0.8335\) nm, close to that of spinel ZnGa\(_2\)O\(_4\) (\(a = b = \sim 0.8335\) nm, JCPDS No. 38-1240). Due to the larger ionic radii of Eu\(^{3+}\) (for sixfold coordination, \(r_{\text{Eu}^{3+}} = 0.0947\) nm and \(r_{\text{Ga}^{3+}} = 0.062\) nm) [25], a larger value of \(a = b = \sim 0.8336\) nm was observed for ZGGO:Cr\(^{3+}\),Eu\(^{3+}\). Profile broadening analysis of the (311) Bragg reflection was conducted by applying the Scherrer equation for average crystallite sizes of \(83 \pm 6\) nm for ZGGO:Cr\(^{3+}\) and ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) samples. In Fig. 1 (up), we also find that the resulting products calcined at 900 °C are the mixture of spinel phase (JCPDS No. 38-1240) and rhombohedral one (JCPDS No. 11-0687), indicating a calcination temperature of \(\geq 1000\) °C is needed for yielding spinel Zn\(_3\)Ga\(_2\)Ge\(_2\)O\(_{10}\) in a phase pure form.

Figure 2 (left) shows the TEM morphology for ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) particles, clearly indicating that they entirely consist of cubic particles, with lateral sizes of \(\sim 80–100\) nm. The sharp corners and the well-resolved lattice fringes suggest their excellent crystallinity, while the spacings of \(\sim 0.29\) nm correspond well to the (220) plain of spinel structured ZnGa\(_2\)O\(_4\) \((d(220) = \sim 0.29\) nm, JCPDS No. 38-1240) (inset in Fig. 2). Because the
particle sizes are close to the average crystallite sizes calculated from the XRD data, the obtained samples may be single crystalline. The SAED analysis (Fig. 2 (right)) further confirmed that the nanoparticles under analysis are of single crystalline. The nanoparticles investigated here are direct solid-state solutions rather than a mechanical mixture. Elemental mapping of Zn, Ga, Ge, Cr, and Eu provides evidence of this solid solution, as revealed in Fig. 3 for ZGGO:Cr$^{3+}$,Eu$^{3+}$. Not only does each particle contain Zn, Ga, Ge, Cr, and Eu, but all elements are evenly distributed among the particles.

Figure 4 shows the excitation spectra of the ZGGO:Cr$^{3+}$ and ZGGO:Cr$^{3+}$,Eu$^{3+}$ powder at room temperature. The excitation spectrum monitored at 697 nm covers a very broad spectral region (from 200 to 650 nm) and consists of four main excitation bands.

![Fig. 8](image_url)
peaking at 273, 328, 410, and 569 nm, respectively. The excitation band at 273 nm is ascribed to the charge transfer band of $\text{O}^2-\text{Ga}^{3+}$ in $\text{ZnGa}_2\text{O}_4$ host, while the later bands originate from the inner transitions of $\text{Cr}^{3+}$, including the 328-nm band originating from the $^4A_2 \rightarrow ^4T_1(\text{f}e)$ transition, the 410-nm band originating from the $^4A_2 \rightarrow ^4T_1(\text{te})$, and the 569-nm band originating from the $^4A_2 \rightarrow ^3T_2(\text{te})$ [19, 20]. Incorporation of Eu$^{3+}$ did not appreciably alter the positions of the PLE bands but significantly increased the intensities of the inner transitions of $\text{Cr}^{3+}$, with $I_{410}/I_{273}$ increasing from 0.18 to 0.56. The above results indicate that Eu$^{3+}$ incorporation is conducive to the visible light excitation. However, the strongest excitation band at 273 nm also revealed that the charge transfer band of $\text{O}^2-\text{Ga}^{3+}$ is the most effective excitation wavelength. Excitation of the powder at 273 nm gave a NIR emission band at 697 nm (Fig. 5) due to the $^2\text{E} \rightarrow ^4\text{A}_2$ transition in distorted $\text{Cr}^{3+}$ ions in gallogermanate, in the absence of Eu$^{3+}$ emission.

The NIR persistent luminescence decay curves of ZGGO:$\text{Cr}^{3+}$ and ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ nanoparticles were monitored at 697 nm after 254 nm UV light illumination (xenon lamp as the light source) for 5 min at room temperature as shown in Fig. 6. The result demonstrates that the NIR persistent luminescence of the ZGGO:$\text{Cr}^{3+}$ sample can last longer than 7200 s and still hold appreciable intensity. The persistent luminescence intensity of ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ increases with the incorporation of Eu$^{3+}$ ion. It is believed that lanthanide ions co-doping with $\text{Cr}^{3+}$ in zinc gallogermanate would yield remarkable NIR persistent luminescence, because of its important role in increasing the amount of anti-site defects which is the responsible for persistent luminescence of $\text{Cr}^{3+}$ in zinc gallogermanate host [1]. On the other hand, NIR persistent luminescence of ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ sample can last longer than that of ZGGO:$\text{Cr}^{3+}$, indicating that Eu$^{3+}$ incorporation can increase the afterglow time. Figure 7 shows NIR afterglow decay images of ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ powders obtained by a Kodak In-Vivo Imaging System FX Pro at different times after stopping UV irradiation, further confirming that the afterglow can last longer than 120 min and keep an intense NIR emission intensity.

For evaluating the performance of phosphor application, especially for high-power applications, thermal stability is a key parameter. To evaluate the thermal quenching behavior of the phosphors in this work, the PL spectra were analyzed at temperatures ranging from 298 to 573 K (Fig. 8). For all samples, elevating the temperature yielded a decrease in the emission intensities at 697 nm. To obtain a more comprehensive picture of the thermal quenching behavior and estimate the value of its activation energy ($E_a$), the Arrhenius equation (Eq. (1)) was employed as follows [26–28]:

$$I_T = \frac{I_0}{1 + c \exp \left(-\frac{E_a}{kT}\right)}$$  \hspace{1cm} (1)

where $I_0$ and $I_T$ are the intensities of the initial and final temperatures, respectively; $c$ is the rate constant; $E_a$ is the activation energy; and $k$ is the Boltzmann constant ($8.629 \times 10^{-5}$ eV K$^{-1}$). Figure 8 shows the ln($I_0 / I_T - 1$) vs 10,000 / $T$ relationship lines for the emission band centered at 697 nm for ZGGO:$\text{Cr}^{3+}$ and ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$. Similar activation energies were calculated: $E_a = 0.23$ eV for ZGGO:$\text{Cr}^{3+}$ and $E_a = 0.25$ eV for ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$. The probability that a nonradiative transition occurs per unit time ($\alpha$) can be defined according to Eq. (2) as follows [29]:

$$\alpha = s \exp \left(-\frac{E_a}{kT}\right)$$  \hspace{1cm} (2)

where $s$ is the frequency factor (s$^{-1}$), $k$ is the Boltzmann constant, and $T$ is the temperature. It can be seen that a lower activation energy ($E_a$) leads to a greater probability ($\alpha$) of a nonradiative transition. Because of the similar activation energy, ZGGO:$\text{Cr}^{3+}$ and ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ exhibited closed thermal stability, indicating Eu$^{3+}$ incorporation did not significantly affect the thermal stability. However, associated phonon side bands (PSBs) centered at 670 nm became dominant at a higher temperature, thus inducing enhanced emission peaks.

We also investigated the PL excitation and emission spectra of the aqueous dispersion of ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ (Fig. 9). Compared to the ZGGO:$\text{Cr}^{3+},\text{Eu}^{3+}$ powder, the...
aqueous dispersion exhibited almost the same profile of the PL excitation and emission curves except the relatively weak excitation intensity at 300 and 600 nm. The weakened intensity is probably due to the quenching effect of the O–H vibration of water.

Hek293T cells were employed here for in vitro imaging test. The as-obtained ZGGO:Cr\(^{3+}\),Eu\(^{3+}\)-NH\(_2\) nanoparticles were dispersed in cell medium (50 mg/mL), which were excited for 10 min by a 254-nm UV lamp and then moved to culture dishes treated for 1 h. Figure 10 (left, red color) shows the cell luminescence imaging collected on a laser scanning confocal microscope in the absence of excitation. The afterglow luminescence signal of the Hek293T cells was still strong enough to be precisely measured after 1 h, though the afterglow luminescence signals gradually decreased over time. For comparison, the cell luminescence imaging was collected on a laser scanning confocal microscope by another mode from the same cells stained with 0.5 mL DAPI dye (right in Fig. 10, simultaneous excitation). The similar imaging signals suggested that the ZGGO:Cr\(^{3+}\),Eu\(^{3+}\)-NH\(_2\) nanoparticles possessed an excellent imaging capacity for cells in vitro.

**Conclusions**

In this work, spinel structured ZGGO:Cr\(^{3+}\) and ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) nanoparticles were synthesized by a sol-gel method in combination with a subsequent reducing atmosphere-free calcinations. The samples were investigated via detailed characterizations by combined techniques of XRD, TEM, STEM, SAED, PLE/PL spectroscopy, and temperature-dependent PL analysis. The nanoparticles with uniform cubic shape and lateral size of ~80–100 nm are of single crystalline and homogeneous solid solution. Excitation of the powder at 273 nm gave a NIR emission band at 697 nm due to the \(^2E \rightarrow ^4A_2\) transition in distorted Cr\(^{3+}\) ions in gallogermanate, in the absence of Eu\(^{3+}\) emission. NIR persistent luminescence of the ZGGO:Cr\(^{3+}\),Eu\(^{3+}\) can last longer than 7200 s and still hold intense intensity. The persistent luminescence intensity of ZGGO:Cr\(^{3+}\) and the afterglow time increase with the incorporation of Eu\(^{3+}\) ion. However, Eu\(^{3+}\) incorporation did not significantly affect the thermal stability. Finally, the as-obtained ZGGO:Cr\(^{3+}\),Eu\(^{3+}\)-NH\(_2\) nanoparticles possessed an excellent imaging capacity for cells in vitro.

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**Availability of Data and Materials**

Not applicable

**Authors’ Contributions**

QQW, SYZ, and ZWL carried out the experiments. QZ and QQW were involved in the results discussion and drafted the manuscript. All the authors have read and approved the final manuscript.

**Competing Interests**

The authors declare that they have no competing interests.

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