1.5 μm Epitaxially Regrown Photonic Crystal Surface Emitting Laser Diode

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Abstract— We present an InP-based epitaxially regrown photonic crystal surface emitting laser diode, lasing in quasi-CW conditions at 1523nm.

Index Terms— Photonic Crystal, Semiconductor growth, Semiconductor lasers, Surface emitting lasers.

I. INTRODUCTION

There has been considerable recent interest in the development of photonic crystal surface emitting lasers [1,2]. A photonic crystal (PC) is formed through a 2D periodic variation in refractive index, and when placed within a laser structure, a surface emitting laser can be realised [3].

PCSELs have shown single mode operation [4,5], low divergence [6], polarisation and beam shape control [7-9], beam steering [10], high power and brightness [1], and coherently coupled arrays [11,12]. There has also been significant work on the simulation of devices [13-16] particularly to optimise output power through PC design [17-19], and more recently simulation results have suggested that speeds of 40 GHz could be achieved from a small, resonator embedded device [20].

Future 5G roll-out requires a corresponding increase in the size of data-centres, requiring longer link lengths, and due to problems in situating them in urban areas, to distributed data-centres, again requiring longer link-lengths approaching those of shorter haul metro networks. In order to service this growing data usage, reliable, low-cost, high speed, lasers operating at 1300 nm and 1550 nm will be required, which may be serviced by InP-based PCSEL devices. Additional applications for such surface emitting InP based devices include free-space communications, LiDAR, fibre-based sensing, etc....

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PCSEL devices have traditionally been realised with wafer fusion [4-8], but more recently epitaxial overgrowth through MBE [21,22] and MOVPE [23-26] has been utilised. This move from wafer fusion to epitaxial regrowth has been critical for achieving high output powers [1,27]. This is because the wafer fusion interface contains regions of discontinuous crystallinity, that result in undesirable defect states. Epitaxial regrowth allows single-crystal PCSEL structures to be realised with encapsulated voids or fully in-filled photonic crystals [28].

In this paper we demonstrate an epitaxially re-grown InP PCSEL operating at 1523 nm at room temperature. The realization of epitaxially regrown PCSELs at this wavelength, as opposed to ones via wafer fusion [29], is a critical step in providing a route to engineer high-power sources with high beam quality.

Figure 1 a) A schematic of the device, with cut-out area showing the photonic crystal pattern and active layer. b) A schematic of the corresponding mode distribution inside the device structure. The epitaxial structure is shown, with the various materials in the key below.

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II. DEVICE DESIGN AND SIMULATION

Figure 1 shows a schematic of our InP based epitaxially regrown PCSEL structure, where the structure consists of, from bottom to top, 3.3 μm of n-doped InP, five 6 nm AlGaInAs quantum wells (separated by 8 nm AlGaInAs barrier layers), a p-doped 243 nm PC layer (consisting of GaInAsP and InP with air containing voids), 1.8 μm of p-doped InP, and capped with a p+ InGaAs cap layer. The PC is a circular atom in a square lattice. Assuming an average refractive index for the PC layer of 3.24, the mode shows an overlap with the photonic crystal of 16.4%.

Figure 2 (a) shows a conventional cross-sectional transmission electron microscope (TEM) image of our regrown photonic crystal grating layer (details later). The bright, central region corresponds to a crystallographically void formed within the etched feature; these are encapsulated laterally by GaInAsP and InP above. Such voids are observed within each of the grating holes. Due to the lattice-matched nature of the structure there is minimal contrast between the GaInAsP layer and adjacent InP layers. Figure 2 b) shows a schematic of the TEM image shown in Fig 2 a). The simulation of band structure is calculated by 2D plane-wave (PW) expansion method [31]. The dotted lines, denoted \( n_{\text{eff},a} \) and \( n_{\text{eff},b} \), show the cross section through the material stack used to calculate the refractive index of the atom, and field materials in the PC, hereafter referred to as region a and b, respectively [32].

Figure 3 shows the simulated photonic band structure for our PC calculated utilizing the structural information from Fig 2, where an r/a of 0.17 (where r/a is the ratio between radius and period of the photonic crystal) is determined, as is the position and height of the void. The k-vector is plotted to 0.2 (2π/λ), which was determined to be the collection angle of the optics used to measure the device (described later). As expected from the symmetry of the PC, four modes are obtained. A splitting of the bands at the Γ-point results in the high density of states from which lasing may occur. The insets in Fig 3 show band splitting of modes A and B. Shown adjacent are the in-plane electric field highlighting that bands C and D are leaky and bands A and B are non-leaky and therefore the likely lasing modes.

III. DEVICE FABRICATION

The devices were fabricated on InP epitaxial wafers, designed to emit around 1550 nm. The epitaxial layer structure of the base planar was as follows (with doping levels in brackets); 3 μm - thick n+ InP buffer layer (1.5 x 10^{18} cm^{-3}), 300 nm n-type InP (1 x 10^{18} cm^{-3}), 15 nm n-type Al_{0.43}Ga_{0.57}In_{0.03}As (1 x 10^{18} cm^{-3}), 15 nm n-type Al_{0.40}Ga_{0.60}In_{0.02}As (6 x 10^{17} cm^{-3}), 12 nm un-doped Al_{0.26}Ga_{0.74}In_{0.02}As, five 60nm-thick Al_{0.26}Ga_{0.74}In_{0.02}As quantum wells with four 8nm – thick Al_{0.26}Ga_{0.74}In_{0.02}As barrier layers, 12 nm of un-doped Al_{0.26}Ga_{0.74}In_{0.02}As, 15 nm p-type Al_{0.38}Ga_{0.62}In_{0.02}As (3 x 10^{17} cm^{-3}), 90 nm p-type Al_{0.43}Ga_{0.57}In_{0.03}As (4 x 10^{17} cm^{-3}), 50 nm p-type InP cladding layer(6 x 10^{17} cm^{-3}), 50 nm p-type InP (8 x 10^{17} cm^{-3}) cladding layer, and a 243 nm - thick p+ Ga_{0.22}In_{0.78}As, 0.45P_{0.52} layer (1 x 10^{18} cm^{-3}).

On these base epitaxy structures, 200 nm of SiO₂ was deposited by plasma enhanced chemical vapour deposition. A square lattice, circular unit cell photonic crystal, with a period of 470 nm, was defined by electron-beam lithography in PMMA and etched into the SiO₂ by reactive ion etching with CHF₃/Ar chemistry. This acts as a hard mask for the etching of the underlying semiconductor. Through this hard mask, the semiconductor is etched to a depth of approximately 170 nm – just above the p-cladding layer atop the AlGaInAs active region with a CHF₃/ H₂ - based inductively coupled plasma etch. SEM images of test structures indicate that the etched InGaAsP PC layer is not modified by the regrowth process. See schematic in Fig 2 b).

The SiO₂ hard mask is then removed and epitaxial regrowth is undertaken. Immediately prior to regrowth, the wafer is uv/ozone cleaned (UVCOS/uv/ozone cleaner) followed by 1 minute in 10:1 buffered HF. Regrowth was performed in an AIXTRON 2400 G2 Planetary MOVPE reactor at 100 mbar, utilising trimethylindium (TMIn) and trimethylgallium.

![Figure 2: a) A TEM image, and b) a schematic of one re-grown circular etched feature in the photonic crystal pattern. The etched GaInAsP can be seen, as well as the regrown InP that fills this etched feature, and a central air-void. The dotted lines in b) denote the cross section from which the effective refractive indices used in further simulations are taken.](image1)

![Figure 3: a) The band structure around Γ-point band edges of a square lattice, circular unit cell photonic crystal, with insets showing degeneracy of the bands and the in plane electric field of each band shown adjacent, the black circles represent the edge of the refractive index contrast in the photonic crystal, and the arrow size represents the intensity of the electric field at each point](image2)
(TMGa) as group-III precursors, and phosphine (PH\textsubscript{3}) and arsine (AsH\textsubscript{3}) as group-V sources. Initially 243 nm of InP was deposited at a growth rate of 10 nm/min at 600 °C. Following this, 1.8 \( \mu \)m of p-doped InP was grown. Regrowth is finished with 25 nm of p-GaInAsP and a 75 nm - thick p+-InGaAs contact layer.

Following re-growth, a 200 \( \mu \)m x 200 \( \mu \)m square mesa is patterned with photolithography and etched in a solution of sulphuric acid and hydrogen peroxide. This etches the regrown material, to a depth of 100 nm. A 200 nm - thick SiO\textsubscript{2} passivation layer is then deposited across the quarter wafer, with contact windows etched into this with a CHF\textsubscript{3} – and Ar – containing reactive ion etch. Following the opening of these contact windows, Ti/Pt/Au is deposited on the top surface of the quarter wafer. A 200 \( \mu \)m square contact with a “lollipop” shaped aperture of 100 nm diameter in the centre is created via a conventional metal lift-off process, the contact shape is illustrated in Fig 1 a). 57% of the device emission area is covered by this contact. A Ni/Au/Ge/Ni/Au n-type contact is then deposited on the bottom side of the quarter wafer, via electron-beam evaporation. This is then annealed at 400 °C for 1 minute. Thick Ti/Au bond-pads are added to the top surface of the quarter wafer, by an electron-beam evaporation and lift-off process.

IV. RESULTS

Devices were measured at 15 °C under quasi-cw conditions using a 10% duty cycle and 10\( \mu \)s pulse width. Surface emission was collected using a NA =0.34 lens, and was focused into a multi-mode fibre-optic cable. Electro-luminescence spectra were measured using an optical spectrum analyser with a resolution of 0.1 nm.

Figure 4 a) shows the current-power characteristics of a typical device showing a threshold current of 640 mA \((J=1.6 \text{ kA/cm}^2)\). An average slope efficiency of \( \sim 0.002 \text{ W/A} \) is obtained, which is low due to the circular symmetry of the PC [33,34] and the masking of the PCSEL emission by the contacts. Figure 4 b) shows the sub-threshold electro-luminescence at 450 mA. Two main peaks are observed at 1512 nm and 1527 nm, attributed to the modification of the spontaneous emission spectrum by the PC. Figures 4 c) and 4 d) show the electroluminescence spectra of the same device at 700 and 900 mA, respectively. A clear lasing peak is observed at 1523 nm in both cases. Figure 5a) shows the EL spectrum of the PCSEL in Fig 4 c) plotted over a narrower wavelength range.

Figure 5 b) shows the simulated optical density of states (ODOS) for our photonic crystal. The 2 peaks are attributed to the 2 band edges of the photonic crystal. This ODOS is calculated by integrating the simulated band structure over the range of k-vectors that are expected to be collected through our measurement system (NA = 0.34). The observed features are in good agreement with the peak position of the long wavelength peak showing excellent agreement with the experimental spectra shown in Fig 5 a), which are the non-leaky modes shown in Fig 3.

V. SUMMARY

We have reported the realisation of an epitaxially regrown PCSEL at 1.5 \( \mu \)m, operating quasi-CW at room temperature opening a new route to surface emitting lasers in InP materials.
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