Foamed concrete (FC) was prepared from raw materials of vanadium tailings (VTs), reservoir sediment (RS), and phosphogypsum (PG). The physicochemical properties of the raw materials were studied by using X-ray diffraction (XRD), X-ray fluorescence (XRF) analysis, and inductively coupled plasma emission spectrometer (ICP-OES). The preparation and properties of FC were investigated by particle size analysis and strength test. The hydration products and microstructure of FC were analyzed by XRD and field emission scanning electron microscopy (FE-SEM). The results show that when the specific surface area (SSA) of VTs is 768 m²·kg⁻¹ and the content is 40%, the products with a compressive strength of 3.56 MPa and density of 619.1 kg·m⁻³ meet the requirements of JG/T 266-2011 standard on FC of grades A06 and C3.0; the main mineral phases in the products are calcium silicate hydrate (C-S-H) gel, ettringite (AFt), and calcite, as well as the residual mineral phases after the system reaction include quartz, orthoclase, mullite, pyrite, and PG.

1. Introduction

Vanadium tailings (VTs) are the tailings produced as the metal vanadium is extracted from stone coal through crushing, calcination, and leaching. VTs mainly encompass SiO₂ and a small amount of Al₂O₃ [1–3]. The production of every 1 t of V₂O₅ from stone coal would generate 150 t of VTs. Usually, VTs are treated by tailings dam stacking in the concentrator. The chemical components of VTs and residual beneficiation reagents produce serious damage to the ecological environment, especially the VTs containing heavy metals, whose loss will harm the whole ecological environment to the people. The accumulation of large quantities of VTs brings economic, environmental, and health problems to many countries [4].

The recycling of tailings is an effective way for mining enterprises to increase economic benefits and improve the eco-environment. Lu et al. and Wang et al. [5, 6] demonstrated that, after being grinded into ultrafine powders, the SiO₂-rich tailings can actively react with alkaline compounds like Ca(OH)₂; the active SiO₂ and Al₂O₃ can react with Ca(OH)₂ and CaSO₄ to produce calcium silicate hydrate (C-S-H) gel, calcium aluminate hydrate (C-A-H), or ettringite (Aft). Therefore, it is feasible to take tailings as the siliceous raw material for the preparation of building materials [7, 8].

Reservoirs are hydraulic engineering structures that regulate water flow and resist floods and droughts. To a certain extent, a reservoir can change the transfer and exchange of material and energy between water bodies. Under the effects of the landform and hydrodynamics of the reservoir, the sediment is gradually deposited at the bottom. If conditions allowed, the sediment will release a large amount of organic matter and heavy metals into the water through biogeochemical processes, thereby affecting the ecosystem of the reservoir.
The reservoir sediment (RS) boasts a high utilization value because it is similar to the soil in terms of properties and rich in organic matter and minerals. Harmless treatment and recycling of RS would be environmentally friendly and resource-efficient. As a byproduct of the phosphorus chemical industry, phosphogypsum (PG) has a cumulative stockpile of more than 600 million t but the comprehensive utilization rate is below 40% [9]. To recycle RS and PG, one of the most important pathways is to prepare them into building materials such as building bricks, lightweight aggregates, and cementing materials [10–14].

Traditional foamed concrete (FC) is a lightweight porous concrete prepared by mixing cementing materials, mineral admixtures, additives, foaming agents, and foam stabilizers at a certain ratio [15, 16]. The advantages of FC include lightweight, good fluidity, strong thermal insulation, and a wide density range (300–1,850 kg·cm⁻³) [17]. This energy-saving green material can replace the traditional clay bricks and thereby save land resources [18].

The use of industrial solid waste to prepare FC has been widely used as a way to absorb solid waste. The foam concrete prepared by solid waste has good performance and wide application. FCs are being prepared with various solid wastes as raw materials, such as slag, fly ash, metakaolin, steel slag, and desulfurized gypsum [19–22]. Domestic and foreign scholars have prepared FCs solely from VTs, RS, or PG. However, there is no report on mixture of VTs, RS, and PG to prepare FC. The composition of VTs, RS, and PG differs greatly from country to country and from region to region. Domestically, there is an urgent need to treat these three types of wastes. Therefore, it is necessary to study whether it is feasible to mix the three materials into FC.

Based on the idea of treating waste with waste, this paper carries out experiments on the preparation of FC by mixing VTs, RS, and PG. The aim is to prepare A06 and C3.0 grade FC that meets the requirements of JG/T 266–2011 Foamed Concrete Industry Standard, making it a new type of lightweight wall material with good thermal insulation performance. The use of VTs, RS, and PG to prepare foamed concrete can achieve the high-value utilization of quartz and other minerals in solid waste. It can provide new ways to improve the comprehensive utilization value and utilization rate of solid waste, save land, save resources, and reduce environmental pollution, so as to provide technical support for constructing a circular economy industrial chain in mining areas. It can effectively alleviate the social contradictions and the pressure on the environment and safety caused by solid waste storage in mining enterprises. On the other hand, the preparation of FC insulation materials meets the low carbon and environmental protection development strategy proposed by our government, representing the development direction of the transformation and upgrading of China’s construction industry in the future. The physicochemical properties of the raw materials were studied by using X-ray diffraction (XRD), X-ray fluorescence analysis (XRF), and inductively coupled plasma emission spectrometer (ICP-OES). The preparation and properties of FC were investigated by particle size analysis and strength test.

The hydration products and microstructure of FC were analyzed by using XRD, field emission scanning electron microscopy (FE-SEM), and X-ray energy spectrum analysis (EDS). The research results provide theoretical and technical supports to the recycling of VTs, RS, and PG.

2. Raw Materials and Experimental Method

2.1. Raw Materials

2.1.1. Vanadium Tailings (VTs). The VTs used in our experiments are the waste slag of acidic extraction of vanadium from stone coal, a kind of vanadium ore. The waste slag is black in appearance, acidic, and 2.61 g·cm⁻³ in density. From the XRD pattern in Figure 1, it can be observed that the VTs mainly encompass minerals like quartz and orthoclase, plus a few anhydrites and pyrites. The XRF data in Table 1 show that SiO₂ is the primary chemical component of the VTs, with a content of 64.21%. About 93.67% of the VTs can be retained by a 0.08 mm square hole sieve. For the VTs, the extraction toxicity of heavy metals meets the requirements in the National Standard GB/T 5085.3-2007 Identification Standards for Hazardous Wastes-Identification for Extraction Toxicity (Table 2).

2.1.2. Reservoir Sediment (RS). The RS used in our experiments has a gray surface, fine particles, and a few impurities like water plants. The water content was tested as 70%. The RS was air-dried, stoved, and calcinated at 1,300°C. The calcined product was quenched with water, stoved, and ground with a ball mill into the raw material. Tables 1 and 2 show the main chemical components and extraction toxicity of the ground RS, respectively. The ground RS is mainly composed of inorganics, including SiO₂ (62.34%), Al₂O₃ (16.41%), Fe₂O₃ (5.29%), CaO+MgO (10.12%), and K₂O+Na₂O (3.20%). The heavy metal contents of Cu, Zn, Pb, and Cd meet the requirements in GB/T 5085.3-2007. The ground RS has a specific surface area (SSA) of 522 m²·kg⁻¹. The mineral components include quartz, mullite, anorthosite, and calcium akermanite (Figure 1(b)). The extraction toxicity of heavy metals meets the requirements in GB/T 5085.3-2007 (Table 2).

2.1.3. Phosphogypsum (PG). Tables 1 and 2 show the main chemical components and extraction toxicity of the PG, respectively. PG exists in the form of gray powders, whose particle size is 5 μm–150 μm and relative density is 2.31 g·cm⁻³. The main mineral components are CaSO₄·2H₂O and CaSO₄·1/2H₂O (Figure 1(c)). The extraction toxicity meets the requirements in the said national standard.

2.1.4. Lime. Table 1 shows the chemical components of lime. The lime can be digested in 14 min at 67°C. Less than 15% can be retained by a 0.08 mm square hole sieve, which meets the requirements in JC/T 621-2009 Lime for silicate building products.
Figure 1: XRD patterns of raw materials: (a) VTs, (b) RS, and (c) PG.

| Materials | SiO$_2$ | Al$_2$O$_3$ | CaO | MgO | Fe$_2$O$_3$ | FeO | Na$_2$O | K$_2$O | SO$_3$ | P$_2$O$_5$ | LOI |
|-----------|---------|-------------|-----|-----|-------------|-----|--------|--------|--------|-----------|-----|
| VTs       | 64.21   | 6.41        | 6.60 | 0.37 | 3.53        | 0.07 | 0.13   | 3.20   | 5.39   | 0.04      | 9.46|
| RS        | 62.34   | 16.41       | 6.81 | 3.31 | 5.29        | 0.17 | 0.58   | 2.62   | —      | 1.84      | 0.34|
| PG        | 14.57   | 0.13        | 33.72| 0.17 | 0.15        | 0.02 | —      | 0.21   | 45.75  | 0.94      | 4.01|
| OPC       | 22.50   | 4.86        | 66.30| 0.83 | 3.43        | 0.21 | 0.24   | 0.31   | —      | 0.02      | 0.12|
| Lime      | 5.45    | 3.85        | 78.76| 3.59 | 1.68        | 0.08 | 0.39   | 1.25   | 0.45   | 0.24      | 3.93|

Table 2: Extraction toxicity of raw materials.

| Materials | Cu (mg/L) | Pb (mg/L) | Zn (mg/L) | Ni (mg/L) | Cd (mg/L) | Cr (mg/L) | As (mg/L) | Hg (mg/L) | pH |
|-----------|-----------|-----------|-----------|-----------|-----------|-----------|-----------|-----------|-----|
| VTs       | 0.028     | 0.42      | 0.33      | 0.0009    | 0.0013    | 0.021     | 0.0056    | 0.000021  | 7.24|
| RS        | 39.241    | 0.24      | 56.57     | 0.0165    | 0.13627   | 0.258     | 0.1673    | 0.00336   | 7.21|
| PGS       | 0.925     | 1.84      | 24.63     | 0.0049    | 0.2818    | 2.556     | 0.1873    | 0.06258   | 3.72|
| Limit value | 100      | 5         | 100       | 5         | 1         | 5         | 5         | 0.1       |     |
2.1.5. Ordinary Portland Cement (OPC). The PO 42.5 OPC was adopted for our experiments. The chemical components of the OPC are shown in Table 1. The initial and final setting time periods are 118 min and 190 min, respectively, which are in line with the requirements in GB 175-2007 Ordinary Portland Cement.

2.1.6. Aluminum Powder (AP). The active aluminum content is 88%, the solid content is 77%, and the foaming rate is 91% in 16 min and 99% in 30 min. 1.5% of the AP can be retained by a 200-mesh sieve. The AP has good water dispersibility and no pellets. Each gram of AP can cover 5,150 cm$^2$ of the water surface if it is arranged continuously as a layer of single particles. The physical properties of the OPC are shown in Table 3.

2.1.7. Other Materials. The foam stabilizer is a specific mixture of oleic acid, triethanolamine, and water at room temperature. After adding the composite foam stabilizer, the suspension performance of Al powder particles is improved; it is in direct contact with water and the gas generation is smooth. At the same time, due to the high-liquid phase content in the pores, the liquid phase contact area is increased and the gas-liquid mass transfer coefficient and reaction speed are increased; so more hydration products are generated, the hydration products are stacked densely and the cementation ability between particles is enhanced, which plays a great role in improving the compressive strength of FC products.

2.2. Experimental Method. The VTs were dried at 105°C in an electro-thermostatic blast drying oven until the water content fell below 0.1%. Then, 5 kg of VTs sample was weighed and grinded in an SM$\phi$500 mm × 500 mm cement test grinder.

The RS was collected, dewatered through deposition, and air-dried in a cool place indoors. After that, the material was dried at 105°C in the oven and calcined in a CD-1700X muffle furnace, with a holding time of 4 h. After calcination, the material was quenched to normal temperature in water. Then, the sample was dried at 105°C in the oven until the water content fell below 0.1% and grinded in the SM$\phi$500 mm × 500 mm cement test grinder.

The PG was collected, screened using an RK/ZS standard sieve shaker, and dried at 105°C in the oven until the water content fell below 0.1%. Next, the material was grinded with an agate grinder into powders that can pass through the −0.074 mm standard sieve.

The ground VTs, RS, and PG were mixed with lime and OPC by a certain ratio (Table 4). The radioactive indices of the dry mixture meet the requirements in the National Standard GB/T6566-2010 Limits of Radionuclides in building materials (Table 5).

Next, 50°C warm water was added to the mixture at the water cement ratio of 0.6, followed by a 90 s-long stirring. Afterward, an amount of AP, equivalent to 0.07% of the dry mixture, was added, followed by 40 s-long stirring. Then, the mixture was poured, foamed, kept still, demolded, and cured into concrete. The dry density and compressive strength of the test blocks were measured. The casting mold is of the size 100 mm × 100 mm × 100 mm. The foaming, standing, and steam curing lasted for 10 h at a 55°C constant temperature curing box before demolding.

2.3. Sample Testing. The particle size distribution of the ground VTs was measured using an Ms2000 laser particle size analyzer. The SSAs of the ground VTs and RS were captured with a QBE-9 automatic SSA tester.

The dry density, compressive strength, and water absorption of the samples were measured in reference to JG/T 266-2011 Foamed Concrete. The measurement of compressive strength was carried out with a WDW-50 computer control electronic universal testing machine, which meets the provisions for GB/T 2611-2007 General Requirements for Testing Machines: the measuring accuracy of the testing machine should be ±1%; the failure load of samples should be greater than 20% and smaller than 80% of the measuring range; and the loading speed should be controlled to 2.0 kN/s.

The chemical composition of raw materials was analyzed with an Axios advanced wavelength dispersive X-ray fluorescence (XRF) spectrometer. The heavy metal contents of the raw materials were analyzed with a Prodigy 7 whole-spectrum direct reading inductively coupled plasma emission spectrometer (ICP-OES). The mineral phases of raw materials and FC were analyzed with a Panaco X-ray diffractometer, with the tube voltage of 40 kV and current of 50 mA. The micromorphology of the FC hydration products was observed with a Carl Zeiss SUPRATM 55 field emission scanning electron microscope.

3. Experimental Results and Discussion

3.1. Effect of VTs Fineness on FC Performance. Figure 2 and Table 6 show the particle distributions of VTs powders grinded for different lengths of time. With the extension of grinding time, the general trends are the decline in particle size and the increase of SSA. As the grinding time increased from 30 min to 60 min, the particle size distribution of VTs is narrowed and gradually converged in the direction of small particle size. As the grinding time increased from 30 min to 60 min, the $D_{10}$ of VTs powder dropped from 2.725 μm to 1.524 μm, i.e., the characteristic particle size reduced by 44.1%; $D_{50}$ dropped from 25.803 μm to 7.852 μm, i.e., the characteristic particle size reduced by 70.1%; and $D_{90}$ dropped from 104.450 μm to 34.209 μm, i.e., the characteristic particle size reduced by 67.3%. At 50 min and 60 min, the SSAs were 768 m$^2$·kg$^{-1}$ and 796 m$^2$·kg$^{-1}$, respectively, the growth of SSA slowed down, and the grinding efficiency dropped. It is not very meaningful to increase the SSA of VTs by further extending the grinding time. Hence, during the early phase of grinding, the VTs particles mainly went through size reduction. With the increase of fineness, the speed of size reduction slowed down. Once the grinding time reached a threshold, further grinding led to the equilibrium...
| Fineness (residue on 80 μm sieve) (%) | Normal consistency (%) | Setting time (min) | Stability | Flexural strength (MPa) | Compressive strength (MPa) |
|-------------------------------------|------------------------|-------------------|-----------|------------------------|--------------------------|
| 22.13                               | 27.2                   | 118               | Final setting | 5.3                    | 32.5                     |
|                                     |                        |                   | 190        | 8.8                    | 59.7                     |

Table 3: Physical properties of OPC.
state, under which the particle size was no longer reduced and the particles even clustered together.

Figure 3 records the effect of fineness on VTs activity. During the experiments, VTs powders of different SSAs (content: 30%) were mixed with PO42.5 OPC. According to the GB/T12957-2005 Test method for activity of industrial wasteslag used as addition to cement, the 28d compressive strength rate of cement mortar was calculated (activity index $R_{28}$). Figure 3 shows the activity index of the ground VTs. It can be observed that, with the increase of SSA, the activity index of the VTs initially declined, reached the valley at the SSA of 475 m²·kg⁻¹, and then gradually rebounded. When the SSA increased to 475 m²·kg⁻¹, the activity index of the VTs peaked at 70.7%, presenting an inflection point. In this case, the SSA increase has little promoting effect on VTs activity. This is consistent with the particle size variation of VTs.

3.2. Effect of VTs Content on FC Performance. Figure 4 shows the effect of different VTs contents on the physical performance of FC. The 28 d compressive strength of FC increased first and then decreased, with the growing content of VTs. When the VTs content was 40%, the compressive strength of FC reached 3.56 MPa and the oven-dry density was 619.1 kg·m⁻³, which satisfy the requirements of JG/T 266-2011 on A06 and C3.0 FC. With the growing content of VTs, the strength of FC started to decline; meanwhile, the oven-dry density of FC gradually increased. At the VTs content of 30%, oven-dry density stood at 596.4 kg·m⁻³; when the VTs content rose to 50%, the oven-dry density climbed up to 652.5 kg·m⁻³. The main reason may be that the content of VTs in the D1 sample of FC is low, the content of active SiO₂ in the raw material system is low, and there are few hydration products produced by active SiO₂ and Ca(OH)₂ in slurry, resulting in too large pores formed by aluminum powder gas generation in the D1 sample and low-dry density and compressive strength of the sample [23]. When the VTs content reached 50%, all the pores of FC were filled up, resulting in the peak oven-dry density. However, under a constant water cement ratio, further growth in VTs content pushed up the consistency of the slurry, dragging down the overall fluidity. During the foaming process, some of the air cannot be released. Hence, the foaming height failed to meet the requirements, leading to a poor visual effect of the FC. To make full use of the VTs, the VTs content was determined as 40% from the perspectives of economy and performance (Table 4).

3.3. Phase Analysis of FC

3.3.1. XRD Analysis. Figure 5 displays the XRD patterns of D3 FC samples cured for 3 h, 3 d, 7 d, and 28 d, respectively. The variation in curing time led to changes in the hydration products of FC. Compared with Figure 1, the 1d curve in Figure 5 contained the characteristic peaks of quartz and orthoclase; the diffraction peak of APT was not observed, indicating that the hydration reaction had not taken place.
yet. With the growth of curing time, the diffraction peaks of all mineral phases started to decline, while the diffraction peak of AFt, a hydration product, came into being. (Q_his is because the cement is hydrated during standard curing and the active particles in the raw materials start to react with Ca(OH)₂, the digestion product of CA, producing hydration products of C-S-H gel and C-A-H crystal; in the presence of PG, the Ca²⁺ and SO₄²⁻ released by PG react with the OH⁻ and AlO₂⁻ in the slurry and the product C-A-H will quickly form the AFt crystal [24]. As can be seen in the XRD patterns of 3 d, 7 d, and 28 d, the diffraction peak of gypsum gradually weakened and the diffraction peak of the AFt crystal was enhanced with the extension of curing time. Besides, steamed bun-shaped peaks appeared at 25°–35°. The reason is that the diffraction peaks are widened due to the production of hydration products with indefinite form and poor crystallization capacity and the presence of small-to-medium particles in FC [25]. The gradual formation of hydration products AFt and C-S-H gel in the early phase of sample curing is the primary cause for the hardening of FC blocks. The accumulation of these products facilitates the strength enhancement of FC in the late phase. Among the phases of FC, calcite appears through carbonization during the preparation and curing processes. In the raw material system, there is a low presence of pyrite of the original VTSs. (Q_his is why no diffraction peak of pyrite was observed in the XRD patterns. After 28 d of standard curing, quartz and orthoclase still existed, but their diffraction peaks were weakened. It can be preliminarily concluded that these minerals have not fully reacted under the conditions of standard curing [26]. (Q_he residual quartz, mullite, orthoclase, calcite, pyrite, and PG of the raw materials partly remain in the FC products as aggregates.

3.3.2. SEM Analysis. Several samples were prepared from D3 FCs cured for 1 d, 3 d, 7 d, and 28 d and subjected to the SEM analysis (zoom in 10000 times). The analysis results of these samples are given in Figures 6–8. Specifically, Figure 6 provides the SEM images of cross-sectional pores in 1 d, 3 d, 7 d, and 28 d FCs; Figure 7 displays the SEM-EDS images of the hydration products of 7 d FC; Figure 8 presents the SEM images inside the pores of 3 d and 28 d FCs.

As shown in Figure 6, the cross-sectional pores of FC provide sufficient space for crystal growth. As a result, with the growth of curing time, more and more crystals are formed and the crystal morphology became increasingly regular. On the surface of the 1 d FC, flocculated aggregates were scattered, which are probably the aggregation of gelatinous tobermorite and rod-shaped AFt crystals [27]. The flocculated aggregates gradually increased in the 3 d FC. On the 7th day of curing, the rod-shaped crystals grew along the cross section, were extended in length, and were inserted into the flocculated aggregates. By the 28 d FC, rod-shaped crystals, through centralized growth, basically covered up the cross-sectional pores, forming crystal aggregates that enhance the support to pore walls [28]. The morphological change of aggregates was apparent in the SEM image enlarged by 10,000 times. In the 1 d FC (Figure 6(a)), the aggregates dispersed on the pore walls and the rod-like crystals were attached to the gel surface; in

| Characteristic particle size | Grinding time (min) |
|-----------------------------|---------------------|
|                             | 30                  |
|                             | 40                  |
|                             | 50                  |
|                             | 60                  |
| 30μm (D₃₀)                  | 2.412               |
| 50μm (D₅₀)                  | 19.793              |
| 90μm (D₉₀)                  | 73.909              |
| SSA (m²·kg⁻¹)               | 475                 |
|                             | 600                 |
|                             | 768                 |
|                             | 796                 |

Table 6: Characteristic particle size distributions and SSAs of VTSs at different grinding time periods.
Figure 5: XRD patterns of FC at different curing ages.

Figure 6: SEM images of cross-sectional pores of FC at different ages (zoom in 10000 times). (a) 1 d, (b) 3 d, (c) 7 d, and (d) 28 d.
Figure 7: SEM-EDS images of hydration products in Figure 6(c).

Figure 8: SEM images inside the pores of FC at different ages (zoom in 10000 times). (a) 3 d. (b) 28 d.
the 3 d FC (Figure 6(b)), there were more such aggregates, which almost cover up the cross-sectional pores; in the 7 d FC (Figure 6(c)), the rod-like crystals in the aggregates grew longitudinally, the gelatinous substances gradually decreased, and the said crystals became more prominent and were inserted into the aggregates. From the SEM-EDS images on the hydration products of 7 d FC, it can be inferred that the gelatinous substance at point R is C-S-H gel, that at point T is AFt, and that at point S is the gelatinous substance mixed with AFt-crystallization nuclei. These phenomena agree well with the XRD analysis. As the curing age increased to 28 d (Figure 6(d)), the hydration reaction continued and the hydration products were crystallized to a much higher degree; the AFt crystals became thicker and crisscrossed the entire cross-sectional pores. The interconnected crystals formed a network, making the system denser. The network can alter the pore structure of FC and reduce the concentration of external stress. This helps to enhance the compressive strength of FC and improve the thermal insulation and heat preservation capacity of FC [29].

As shown in Figure 8, the cross section inside the pores of FC is too small for crystal growth. Most crystals developed in capillary pores, which greatly increased the structural density of pore walls. With the extension of curing time, the number of rod-like crystals was on the rise. From the images of 3 d and 28 d FCs (magnified by 10,000 times), there were clear pores in the 3 d FC (Figure 8(a)); by contrast, the pores in the 28 d FC (Figure 8(b)) were basically filled up by the cross-growing rod-like crystals; the enhanced density enables the pore walls to resist greater pressure from the outside (Figure 8(b)).

4. Conclusions

(1) From raw materials like VTs, RS, and PG, this paper successfully prepares the A06 and C3.0 FC, which meets the strength required by the industry standard JG/T 266-2011. The total utilization rate of solid waste is as high as 82%. This is positive for expanding the sources of raw materials for FC, promoting the comprehensive use of solid waste, and protecting the environment.

(2) When the SSA of VTs is 768 m²·kg⁻¹, the mass ratio of materials of VTs is 40%, RS 34%, CA 5%, OPC 13%, PG 8%, and an amount of AP equivalent to 0.07% of the total weight of the dry mixture; the water cement ratio is 0.6; the water temperature is 50°C; the curing temperature is 55°C; the compressive strength of FC sample is 3.56 MPa; and the oven-dry density is 619.1 kg·m⁻³.

(3) XRD and SEM analyses show that, under standard curing conditions, the hydration of FC mainly produces C-S-H gel and AFt; calcite appears through carbonization during the preparation and curing processes; and the residual quartz, mullite, orthoclase, calcite, pyrite, and PG of the raw materials partly remain in the FC products as aggregates.

Data Availability
The data used to support the findings of this study are available from the corresponding author upon request.

Conflicts of Interest
The authors declare that they have no conflicts of interest.

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