Anode Properties of LaSi$_2$/Si Composite Thick-Film Electrodes for Lithium Secondary Batteries

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Abstract. LaSi$_2$ and LaSi$_2$/Si composite as active materials for the negative electrode of lithium-ion battery were synthesized by mechanical alloying. Furthermore, thick-film electrodes prepared with a gas-deposition method by using these material powders, and their electrode performances were investigated. The LaSi$_2$ GD-film electrode exhibited superb cycle stability, where more than 70% of the initial capacity was maintained for a period of 1000 cycles, though the initial capacity was only about 40 mA h g$^{-1}$. As for the LaSi$_2$/Si composite electrodes, the original high discharge capacity of Si was retained even after several hundred cycles. The capacity after 300 cycles was 500 mA h g$^{-1}$, which is larger than the theoretical capacity of graphite electrode practically used. Thus, we succeeded in developing the new composite electrode with both high discharge capacity of Si and good cyclability of LaSi$_2$.

1. Introduction

It is considered that the lithium rechargeable batteries remain the most useful types of rechargeable batteries. However, since ever higher capacities are intensively pursued, metal-based materials are actively being researched as high-capacity substitutes for carbon-based materials[1-20].

Si is well known to have the highest capacity as an anode for the Li-ion battery, whereas it shows poor cycle stability owing to significant volume changes during lithium insertion-extraction, resulting pulverization of the particle[1,2]. To solve the problem, we focused on rare earth silicides, such as LaSi$_2$, because they would exhibit good cycle life performance due to their large thermodynamic stabilities which must suppress the pulverization.

On the other hand, we have discovered that the cyclability of the electrode is remarkably improved if the electrode is prepared by using a gas-deposition (GD) method whereby the raw material powder placed in a guide tube is turned into aerosol by feeding it into an evacuated chamber together with argon gas, and a thick-film is created by projecting it through a nozzle on the substrate at high speed[9-11]. This method is expected to confer various advantages, including the production of electrodes exhibiting a strong adhesion between the particles of the active material as well as between the particles and the electron collector and so on. Therefore, we prepared thick film electrodes of rare earth silicides by using the GD method, and estimated their charge-discharge characteristics.

In order to prepare electrodes which combine the advantages of both LaSi$_2$ and Si, we considered whether it would be possible to synthesize a composite material consisting of Si particles with LaSi$_2$. In this research, the LaSi$_2$ and Si composite powder was synthesized by using a mechanical alloying
(MA) method. Then, we used the resulting material as the source material for the synthesis of a thick-film electrode with the GD method. The electrochemical characteristics of the electrodes obtained were investigated.

2. Experimental details
A procedure of synthesis of rare earth silicide, LaSi₂, as an active material is as follows. A mixture of elemental La chip and Si powder was put in a stainless steel vessel together with balls. The Si/La atomic ratio was 2.0, and the weight ratio of the balls to the sample was about 15:1. The vessel used was sealed with an O-ring to keep an atmosphere of dry argon gas. Milling (MA) was done using a high-energy planetary ball mill at 300 r.p.m and at room temperature. As for the synthesis of LaSi₂/Si composite powder, we used an excess amount of Si in comparison with the stoichiometric ratio of LaSi₂ and used MA processing for both La and Si.

Figure 1 illustrates a schematic diagram of apparatus for gas-deposition and the conditions of deposition. The resulting powdered active materials, which are placed inside the tube, are turned into aerosol as they are carried into the evacuated chamber together with the carrier gas. After this, they are ejected at high speed through a nozzle and projected onto the substrate, i.e. 20 µm thick Cu foil, to form the thick-film electrode. A photograph of LaSi₂ thick-film electrode obtained is shown in Figure 2. The thickness of the film was not uniform but was 2~4 µm.

The electrochemical performance of each GD-film electrode was estimated with a galvanostat by using a three-electrode cell in 1 M LiClO₄/PC(Propylene carbonate). Li metal sheets were used for both the counter and the reference electrode. The measurements were carried out at 0.1 mA for both charge and discharge at 303 K. The cycling tests were conducted for voltages in the range of 0.005 to 2.0 V.

3. Results and Discussion
Figure 3 represents the changes in the discharge (Li-extraction) capacity of LaSi₂ thick-film electrode with respect to the number of cycles. The initial capacity of LaSi₂ was only about 40 mA h g⁻¹, and it was found that the large capacity of elemental Si is not manifested in this silicide, as expected. However, the LaSi₂ electrode exhibited superb cycle stability, where more than 70% of the initial capacity was maintained for a period of 1000 cycles. Although Si has great potential capacity, it is unfavorable for the charge-discharge cycle stability. In order to prepare electrodes which combine the
advantages of both LaSi$_2$ and Si, we considered whether it would be possible to synthesize a composite material consisting of Si particles with LaSi$_2$[11].

As for the processing of the powdered materials which were used as sources for preparing the composite LaSi$_2$/Si electrodes, LaSi$_2$ was prepared first with the MA method, after which Si powder was added and the mixture was stirred together lightly in a mortar. However, we were unable to obtain good performance for the electrode produced in this way. Therefore, after this, rather than following the stoichiometric ratios of LaSi$_2$, we used an excess amount of Si and used MA processing for both La and Si. As a result, compositing was realized at the level of individual particles.

Figure 4 shows the XRD pattern of the composite material prepared with the latter method. Since the patterns for both LaSi$_2$ and Si were confirmed, it was clear that the target composite material was successfully synthesized.

The changes in the discharge capacity following the increase in the number of cycles of the LaSi$_2$/Si composite GD-film electrodes are shown in Figure 5. As for the pristine Si thick-film electrode, the capacity suddenly decayed during 50 charge-discharge cycles. On the other hand, for the LaSi$_2$/Si composite electrodes, the originally high discharge capacity of Si was retained even after several hundred cycles. The capacity after 300 cycles was 500 mA h g$^{-1}$, which is larger than the theoretical capacity of graphite electrode practically used.

Figure 6 displays the charge-discharge (Lithium insertion-extraction) curves of the first cycle for the LaSi$_2$/Si composite GD-film electrodes to discuss the charge-discharge reaction. In all the electrodes, the plateau was observed at 0.4 V on the discharge curves, which corresponds to be due to the following reaction[1,2]. Thus, it was clear from the matching of the charge-discharge potential that it is mainly Si that acts as the active material for these composite electrodes.

\[
\text{Si} + 4.4 \text{Li}^+ + 4.4 \text{e}^- \leftrightarrow \text{Li}_{4.4}\text{Si}
\]

As shown in Figure 7, among the composite electrodes, the one in which the ratio of LaSi$_2$:Si was 7:3 exhibited a large capacity of more than 900 mA h g$^{-1}$ at the first cycle, and ca. 400 mA h g$^{-1}$ even
at the 1000th cycle. Thus, we succeeded in developing the new electrode with both high discharge capacity of Si and good cyclability of the rare earth silicide. Their characteristics make them suitable for application as electrodes in embedded batteries for next-generation electric cars.

Figure 8 gives the result of changing the vertical axis in the Figure 7 from discharge capacity to Coulomb efficiency. The declination of the curves represents the peeling or detachment of the active material from the current collector (Cu foil), and it was found that the declination was remarkably suppressed as the amount of LaSi$_2$ became larger. Based on the fact that the composite electrode prepared in this way exhibited superior cyclability, we inferred that the LaSi$_2$ phase surrounds the Si phase in individual composite particles and acts as a matrix which reduces the stress generated by the volumetric changes occurring in Si during the charge-discharge process.

![Figure 7](image1.png)  
**Figure 7** Changes in the discharge capacity of pristine Si and LaSi$_2$/Si composite GD-film electrodes until 1000th cycle.

![Figure 8](image2.png)  
**Figure 8** Changes in the Coulomb efficiency of pristine Si and LaSi$_2$/Si composite GD-film electrodes with the number of cycles.

![Figure 9](image3.png)  
**Figure 9** Surface SEM images of pristine Si and LaSi$_2$/Si composite GD-film electrodes.

Figure 9 displays surface SEM images of the pristine Si and LaSi$_2$/Si composite GD-film electrodes. In the composite electrode with 50 wt% Si content (Figure 9(b)), the existence of the Si particles was obviously confirmed, so that the electrode exhibited poor cyclability. As for the composite electrode with 30 wt% Si content (Figure 9(c)), it was found that Si particles were homogeneously distributed in the composite electrode. At this optimum composition of LaSi$_2$ and Si in the composite, each Si
particle was surrounded with LaSi$_2$ layer, and the stress generated in the Si particle at Li insertion-extraction was considered to be released by the LaSi$_2$ layer because LaSi$_2$ is more ductile than Si.

4. Conclusions

We obtained the following information with respect to the anode characteristics of rechargeable lithium batteries comprising thick-film electrodes prepared with the GD method by using LaSi$_2$ or LaSi$_2$/Si composite powder as the source material. The LaSi$_2$ electrode exhibited superior cycle stability, where the initial discharge capacity was maintained over 1000 cycles, though its capacity was not enough. We succeeded in developing the new composite electrode with both high discharge capacity of Si and good cyclability of a rare earth silicide LaSi$_2$. We infer that the reason for the improvement in the cyclability results from the fact that LaSi$_2$ reduces the stress generated by the immense volumetric changes occurring in the Si particles.

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References

[1] Chan C K, Peng H, Liu G, Mcilwrath K, Zhang X F, Huggins R A and Cui Y 2008 Nature Nanotechnology 3 31-5.
[2] Beattie S D, Larcher D, Morcrette M, Simon B and Tarascon J M 2008 J. Electrochem. Soc. 155 A158-63.
[3] Yang X, Wen Z, Huang S, Zhu X and Zhang X 2006 Solid State Ionics 177 2807-10.
[4] Ng S H, Wang J, Konstantinov K, Wexler D, Chew S Y, Guo Z P and Liu H K 2007 J. Power Sources 174 823-7.
[5] Dimov N, Kugino S and Yoshio M 2003 Electrochim. Acta 48 1579-87.
[6] Park M S, Lee Y J, Han Y S and Lee J Y 2006 Mater. Lett. 60 3079-83.
[7] Kang Y M, Lee S M, Kim S J, Jeong G J, Sung M S, Choi W U and Kim S S 2007 Electrochem. Commun. 9 959-64.
[8] Sakaguchi H, Toda T and Esaka T 2005 Electrochemistry 73 505-8.
[9] Sakaguchi H, Toda T, Nagao Y and Esaka T 2007 Electrochem. Solid-State Lett. 10 J146-9.
[10] Takai S, Sakaguchi H, Tanaka K, Nagao Y and Esaka T 2008 Electrochemistry 76 293-6.
[11] Iida T, Hirono T, Shibamura N and Sakaguchi H 2008 Electrochemistry 9 644-8.
[12] Guo B, Shu J, Tang K, Bai Y, Wang Z and Chen L 2008 J. Power Sources 177 205-10.
[13] Hassoun J, Mulas G, Panero S and Scrosati B 2007 Electrochem. Commun. 9 2075-81.
[14] Todd A D W, Mar R E and Dahn J R 2007 J. Electrochem. Soc. 154 A597-604.
[15] Hassoun J, Panero S, Taberna P L and Scrosati B 2007 Adv. Mater. 19 1632-5.
[16] Mukaibo H, Momma T, Mohamedi M and Osaka T 2005 J. Electrochem. Soc. 152 A560-5.
[17] Matsuno S, Kohno T, Takami N, Kawashima F and Sawa T 2005 Electrochem. Solid-State Lett. 8 A234-6.
[18] Trifonova A, Wachtlera M, Wagner M R, Schroettner H, Mitterbauer C, Hofer F, Möller K -C, Winter M and Besenhard J O 2004 Solid State Ionics 168 51-9.
[19] Wang L, Kitamura S, Obata K, Tanase S and Sakai T 2005 J. Power Sources 141 286-92.
[20] Stjerndahl M, Bryngelsson H, Gustafsson T, Vaughey J T, Thackeray M M and Edström K 2007 Electrochim. Acta 52 4947-55.