Moiré and generalized Wigner crystal states in WSe$_2$/WS$_2$ moiré superlattices

Moiré superlattices can be used to engineer strongly correlated electronic states in two-dimensional van der Waals heterostructures, as recently demonstrated in the correlated insulating and superconducting states observed in magic-angle twisted-bilayer graphene and ABC trilayer graphene/boron nitride moiré superlattices. Transition metal dichalcogenide moiré heterostructures provide another model system for the study of correlated quantum phenomena because of their strong light–matter interactions and large spin–orbit coupling. However, experimental observation of correlated insulating states in this system is challenging with traditional transport techniques. Here we report the optical detection of strongly correlated phases in semiconducting WSe$_2$/WS$_2$ moiré superlattices. We use a sensitive optical detection technique and reveal a Mott insulator state at one hole per superlattice site and surprising insulating phases at 1/3 and 2/3 filling of the superlattice, which we assign to generalized Wigner crystallization on the underlying lattice. Furthermore, the spin–valley optical selection rules of transition metal dichalcogenide heterostructures allow us to optically create and investigate low-energy excited spin states in the Mott insulator. We measure a very long spin relaxation lifetime of many microseconds in the Mott insulating state, orders of magnitude longer than that of charge excitations. Our studies highlight the value of using moiré superlattices beyond graphene to explore correlated physics.
temperatures of up to 45 K and has an estimated Mott–Hubbard gap of about 10 meV, an order of magnitude larger than that in graphene moiré systems. Surprisingly, we also observe additional insulating states from generalized Wigner crystallization at fractional fillings of $n = 1/3n_0$ and $n = 2/3n_0$. The emergence of these generalized Wigner crystal states necessitates an extended Hubbard model with not only on-site (short-range) but also inter-site (long-range) interactions. In addition, the strong light–matter interaction and unique spin–valley selection rules of TMD moiré heterostructures allow us to circularly create and detect different elementary excitations associated with their strongly correlated ground states. We use circularly polarized light to generate a low-energy pure-spin excitation, and we demonstrate an increased spin lifetime at the Mott insulating state.

We investigate correlated states in a TMD heterostructure using the novel optically detected resistance and capacitance (ODRC) technique. The large semiconductor bandgap in TMDs leads to the formation of Schottky barriers at metal–TMD junctions and correspondingly large contact resistance. This large contact resistance often hampers direct electrical-transport measurements in TMD heterostructures, particularly for low carrier doping and at low temperatures. Our optical detection scheme avoids this difficulty. For the ODRC measurements, we design a special device configuration with two regions (Fig. 1b): one half of the device has a local graphite top gate (region 1), and the other half does not (region 2). We vary the d.c. voltage on the local top gate ($V_{\text{top}}$) to continuously control the carrier doping in region 1, where the charge injection occurs with a time constant of about 1 s. We then add an a.c. excitation voltage ($\Delta V$) to the local top gate. For excitation frequencies higher than 10 Hz, the electrical contact is effectively frozen and the TMD heterostructure is floated electrically (see Methods). In this case, the a.c. excitation voltage leads only to charge redistribution between region 1 and region 2, with no total charge change, and the charge redistribution dynamics depends on the quantum capacitance and resistance in the moiré system. We detect the resulting change of carrier concentration in region 2 ($\Delta n_0$) optically, through the induced change in optical contrast $\Delta \alpha C$ at the intralayer exciton resonance (see Methods, Extended Data Fig. 1). The global graphite back gate is used to set the d.c. doping level of region 2 to optimize the exciton optical response to doping changes.

The a.c. electrical transport in the TMD heterostructure can be modelled by an effective RC circuit, shown in Fig. 1c. Here $C_1$ and $C_2$ are the geometric capacitances between the TMD and the top and bottom gates in region 1, respectively, and $C_3$ is the TMD–bottom gate capacitance in region 2. These geometric capacitances $C_i$ ($i = 1, 2, B$) are set by $C_i = \varepsilon_i\varepsilon_0d_i/A_i$, where $\varepsilon_i$ is the dielectric constant of the gate dielectric, and $A_i$ and $d_i$ denote the relevant capacitor area and separation, respectively. The parameters to be measured are $C_2$ and $R$, which correspond to the doping-dependent quantum capacitance and resistance of the moiré superlattice in region 1, respectively. The induced optical contrast change $\Delta \alpha C$ in region 2 upon an a.c. capacitive excitation $\Delta V$ in region 1 can be obtained from the effective circuit model (see Supplementary Information) as

$$\Delta \alpha C = \alpha \Delta n_0 = \frac{\alpha}{A_2}\frac{\Delta V}{C_1 + C_2/\varepsilon_{\text{eff}}} + \frac{1}{\omega R}$$

(1)
Fig. 2 | Doping-dependent resistance and capacitance probed by ODRC. a, ODRC signal at 1 kHz (grey) and 30 kHz (black) from charge-neutral to moderate hole doping. Strong gap-like features are observed at hole doping levels of \( n = n_0/3 \) (orange dashed line), \( n = 2n_0/3 \) (green dashed line) and \( n = n_0 \) (blue dashed line). The purple dashed line corresponds to \( n = 0 \). b–d, Capacitance \( C_{\text{eff}} \) (b) and resistance (c) of region 1. Grey curves are extracted from the data in a, and black dots are extracted from the frequency-dependent optical detection responsivity in region 2, which is constant for the fixed bottom gate voltage used in our study. The frequency-dependent optical detection responsivity in region 2, which is constant for the fixed bottom gate voltage used in our study.

Here \( \omega \) is the excitation frequency and \( \alpha = \Delta OC/\Delta \tilde{n} \) is the optical detection responsivity in region 2, which is constant for the fixed bottom gate voltage used in our study. The frequency-dependent optical signal \( \Delta OC(\omega) \) allows us to extract the values of both \( C_0 \) and \( R \): at low excitation frequencies the resistance is negligible, so the optical signal probes the quantum capacitance \( C_0 \), which is proportional to the density of states of the moiré heterostructure. At high modulation frequencies, both \( C_0 \) and \( R \) contribute to the optical signal.

We focus our study on WSe\(_2\)/WS\(_2\) heterostructures with near-zero twist angles that have a moiré superlattice with a period of about 8 nm owing to the ~4% lattice mismatch between the WS\(_2\) and WSe\(_2\) monolayers. Figure 1d shows a schematic of device D1: few-layer graphene is used for the gates and contact to the TMD layers, and hexagonal boron nitride (hBN) is used at the top and bottom gate dielectrics (\( \varepsilon_r = 4.2 \); see Methods and ref. 25 for fabrication details). Figure 1e shows an optical microscopy image of the final device, with contours highlighting the WS\(_2\) and WSe\(_2\) layers and the local graphite top gate. To verify the presence of the moiré superlattice, we examine the optical absorption spectrum of the heterostructure (Fig. 1f). The spectrum shows clear splitting of the WSe\(_2\) A exciton, which is a signature of the moiré superlattice in the heterostructure.

Figure 2a shows the ODRC signals as a function of the hole doping of the WSe\(_2\)/WS\(_2\) moiré superlattice in region 1. We use an a.c. excitation voltage with the peak-to-peak amplitude of 10 mV at 1 kHz and 30 kHz. When region 1 is charge-neutral (\( V_{\text{top}} = 0.2 \) V), the \( \Delta OC \) signal is small because no carriers are available to redistribute in the bandgap of WSe\(_2\). When region 1 is hole-doped (\( V_{\text{top}} < 0.2 \) V), charge redistribution occurs, leading to a large increase in the signal. Interestingly, we observe a strong gap-like feature at \(-1 \) V (blue dashed line in Fig. 2a). From a capacitance model, we estimate the corresponding hole concentration to be \( 1.86 \times 10^{12} \) cm\(^{-2}\), which matches well with a density of one hole per moiré unit cell (\( n_0 = 1.88 \times 10^{12} \) cm\(^{-2}\); see Methods). We also observe two sharp dips at \(-0.2 \) V and \(-0.6 \) V (orange and green dashed lines in Fig. 2a), which correspond to hole concentrations of \( n = n_0/3 \) and \( n = 2n_0/3 \), respectively. Additionally, a broad, weaker feature is observed at \(-2.25 \) V, which corresponds to \( n = 2n_0 \). These features become stronger at the higher excitation frequency of 30 kHz. ODRC signals for additional aligned heterostructures are shown in Extended Data Fig. 5.

We extract numerical values for the doping-dependent \( C_{\text{eff}} \) and \( R \) of the moiré heterostructure based on the effective a.c. circuit model and equation (I). We plot \( C_{\text{eff}} \) and \( R \) as a function of carrier doping in Fig. 2b, c (grey lines). An optical responsivity of \( \alpha = 1.4 \times 10^{-14} \) cm\(^2\) is chosen so that \( \frac{1}{C_{\text{eff}}} = \frac{1}{C_0} + \frac{1}{C_G} \) at high doping, where the quantum capacitance is much larger than the geometric capacitances and has negligible contribution. At \( n = n_0 \), \( n = n_0/3 \) and \( n = 2n_0/3, C_{\text{eff}} \) decreases, whereas the geometric capacitances remain unchanged (Fig. 2b). This decrease in \( C_{\text{eff}} \) is due to the much smaller quantum capacitance \( C_0 \), which results from the greatly reduced density of states at these fillings. At the same time, the electrical resistance shows marked increases at \( n = n_0, n = n_0/3 \) and \( n = 2n_0/3 \) (Fig. 2c). The simultaneous reduction of the density of...
states and large increase of the resistance indicate the emergence of insulating states at these fillings.

To test our effective circuit model quantitatively, we measure the frequency dependence of the ODRC signal at several representative hole-doping densities. Figure 2d displays the experimental results (symbols). We observe a clear signal fall-off with increasing frequency, and the data can be reproduced by the circuit model predictions (solid lines). The effective capacitance and resistance at these fillings extracted from the frequency dependence of the ODRC signal (black dots in Fig. 2b, c) agree well with the values extracted directly from the data in Fig. 2a.

Our results show that the WSe₂/WS₂ moiré heterostructure hosts insulating states with reduced density of states and increased resistance at \( n = n_0, n = 2n_0/3 \) and \( n = 2n_0/3 \). These features are completely absent in large-twist-angle WSe₂/WS₂ heterostructures (see Methods, Extended Data Fig. 6) and only emerge in the moiré superlattices. The insulating state at \( n = n_0 \) is typically assigned to a Mott insulator but may also be considered to be an interaction-driven Wigner crystal state or a charge-transfer insulator (Fig. 2e). This state corresponds to half-filling of the moiré miniband because the TMD heterostructure has a degeneracy of 2 from spin–valley locking. Similar correlated insulating states have also been observed at \( n = n_0 \) in twisted bilayer graphene and ABC trilayer graphene/boron nitride moiré superlattices.

On the other hand, the observation of insulating states at \( n = n_0/3 \) and \( n = 2n_0/3 \) is surprising. Insulating states at fractional filling of the lattice sites have not been observed in other moiré superlattice systems and cannot be described as a Mott insulator or by a Hubbard model with only on-site repulsive interactions. We hypothesize that these insulating states at \( n = n_0/3 \) and \( n = 2n_0/3 \) correspond to generalized Wigner crystallization of holes in the TMD moiré superlattice. Figure 2e illustrates the real-space configurations of the generalized Wigner crystal states, where holes try to avoid not only double occupation in one site, but also simultaneous occupation of adjacent sites. There are three degenerate Wigner crystallization configurations. The TMD moiré system spontaneously breaks the lattice translational symmetry owing to electron–electron interactions and condenses to one specific configuration with \( \sqrt{3} \times \sqrt{3} \) charge density wave pattern. The emergence of these generalized Wigner crystal states suggests that the inter-site (long range) interaction energy is larger than the moiré miniband bandwidth, confirming the very strong correlation in the TMD moiré heterostructure.

We perform ODRC measurements of the doping-dependent quantum capacitance and resistance of the TMD moiré superlattices at different temperatures. Figure 3a shows the extracted resistance for temperatures from 3 K to 70 K. The resistance peaks of the Mott insulator and generalized Wigner crystal states are observable up to temperatures of 45 K and 10 K, respectively. We estimate the Mott−Hubbard gap to be \( \Delta = 10 \text{ meV} \) by fitting the resistance with a thermal-activation function, \( \exp[-\Delta/(2k_B T)] \) (\( k_B \) Boltzmann constant; \( T \) temperature), for the Mott insulator state at \( n = n_0 \) (black dashed line in Fig. 3b). Owing to the limited range exhibiting thermal-activation behaviour, the estimated Mott gap has relatively large uncertainty. It is difficult to estimate the size of the insulating gaps of the generalized Wigner crystal states from the experimental data, but they are probably 5–10 times smaller than the Mott insulator gap according to the temperature at which the generalized Wigner crystal signatures disappear.

The strong electron correlation and light–matter interaction in the heterostructure provides unique opportunities to optically investigate excited states from the correlated phases, such as low-energy charge and spin excitations. Charge excitations in Mott insulator systems have been intensively studied, featuring ultrafast decay dynamics (typically a few picoseconds) from the holon−doublon recombination process. On the other hand, the dynamics of pure spin excitations are difficult to explore. Here we directly measure the doping-dependent decay of a pure spin excitation by taking advantage of the unique spin−valley selection rules in the TMD heterostructure. We use the pump−probe scheme described in refs. to generate and probe the spin excitation
in the moiré heterostructure at 20 K. Specifically, a circularly polarized pump excitation is employed to selectively excite K-valley excitons composed of spin-up holes and electrons. The relaxation of the spin-polarized electrons and holes within about 100 ns results in a residual spin polarization in the lower Hubbard band of the Mott insulator, as illustrated in Fig. 4a. We probe the evolution of the residual spin polarization through the pump-induced circular dichroic signal and the charge population through the pump-induced change in the total absorption of the probe beam. Figure 4b shows the time evolution of the spin population at different hole densities. The doping-dependent spin lifetime, represented by blue symbols in Fig. 4c, shows a prominent increase at the Mott insulator state \((n = n_c)\) and reaches more than 8 μs. By contrast, the lifetime of charge excitations (black symbols) is orders of magnitude shorter. The long-lived spin excitations from the Mott insulator state can provide important information about its spin configuration. It has been proposed that the Mott insulator state in the WSe\(_2\)/WS, moiré superlattice can host intriguing spin states, such as the quantum spin liquid.\(^{12,13}\) However, further theoretical studies will be required to understand experimentally observed spin dynamics in the Mott insulating state, which is beyond the scope of this paper.

Our results demonstrate that TMD moiré heterostructures can host novel quantum correlated phases and offer an attractive platform for probing excited-state and non-equilibrium dynamics of the correlated phases owing to a unique combination of highly correlated electrons, strong light–matter interactions and a large spin–orbit effect in the system.

**Online content**

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Methods

ODRC measurements
A function generator (Rigol 1022Z) is used to generate the top-gate voltage, which consists of a d.c. offset, $V_{\text{top}}$, and a small a.c. modulation, $\Delta V$. A voltage source (Keithley 2400) is used for the back-gate voltage. A laser diode with centre energy of 1.66 eV serves as the probe light. The diode energy is fine-tuned using a thermoelectric cooler so that the probe energy is resonant with the lowest-energy WSe$_2$ A exciton absorption peak in region 2. The reflected probe light is collected with an avalanche photodiode (Thorlabs APD 410A) and then analysed using a lock-in amplifier that is locked to the function generator output.

WSe$_2$ A exciton absorption in region 2
The lowest-energy WSe$_2$ A exciton in region 2 is used to measure the electrical properties of region 1 as the local top gate is tuned. To be a reliable probe, the exciton in region 2 must respond only to charge redistribution due to the modulation voltage $\Delta V$, but not to the d.c. bias applied to the local top gate, $V_{\text{top}}$. We measure the optical spectrum of the lowest-energy WSe$_2$ A exciton (1.6–1.74 eV) while sweeping $V_{\text{top}}$ from 0.5 V to −3.5 V (Extended Data Fig. 1a). We record the spectrum 15 s after changing $V_{\text{top}}$ to ensure that the contact injects charge. We observe almost no change in the spectrum. Therefore, the static local top gate does not influence the hole concentration in region 2, and it remains a stable probe for all $V_{\text{top}}$ values used.

We measure the absorption spectrum as a function of the carrier density in region 2 by varying the global back-gate voltage. Extended Data Fig. 1b shows the absorption spectra when the back-gate voltage is tuned in a small range around −1 V. For the ODRC measurement, the modulation voltage $\Delta V$ is typically set to 10–25 mV, so the redistributed charge corresponds to a small back-gate voltage change of <25 mV. Within this range, the exciton resonance (1.673 eV) shows monotonic and linear change with carrier concentration. Therefore, the optical detection responsivity $\alpha$ is a constant for our choice of back-gate voltage.

We estimate an optical detection responsivity $\alpha = 1.4 \times 10^{-12} \text{ cm}^2$. The noise of the ODRC signal is -2 × 10$^{-4}$ in our lock-in measurement for 3 s averaging time. This allows us to detect a carrier density change in region 2 as small as 10$^6$ cm$^{-2}$ with optical detection.

Low-frequency behaviour of ODRC signal
Extended Data Figure 2 shows the frequency-dependent ODRC signal at $V_{\text{top}} = −1.6$ V (that is, away from any features) for modulation frequencies between 0.05 Hz and 137 Hz. The carrier injection through the graphite contact has a characteristic time constant of ~1 s. At the lowest modulation frequency (0.05 Hz), the graphite contact can efficiently inject charge in response to $\Delta V$. As a result, the carrier density in region 2 remains constant and the overall ODRC signal is negligible. At 1 Hz, the ODRC signal is partially reduced compared with higher-frequency responses because the contact can inject some charge in response to $\Delta V$. For frequencies higher than ~10 Hz, the contact becomes frozen. As a result, the heterostructure is effectively floated and the ODRC signal reaches its typical low-frequency value. We also note that the ODRC signal is linear with $\Delta V$.

Heterostructure preparation for optical measurements
We use a dry-transfer method with a polyethylene terephthalate (PET) stamp to fabricate the WSe$_2$/WS$_2$ heterostructures. Monolayer WSe$_2$, monolayer WS$_2$, few-layer graphene and thin hBN flakes are first exfoliated onto Si substrates with a 90-nm-thick SiO$_2$ layer. For aligned heterostructures, we use polarization-dependent second-harmonic generation (SHG) to determine the crystal axes of WS$_2$ and WSe$_2$. We then use a PET stamp to pick up the few-layer graphene top gate, top hBN flake, the WS$_2$ monolayer, the WSe$_2$ monolayer, the few-layer-graphene contact, the bottom hBN flake and the few-layer graphene back gate in sequence. Between picking up WS$_2$ and WSe$_2$, we adjust the angle of the PET stamp to ensure a near-zero twist angle between the flakes. The PET stamp with the above heterostructure is then stamped onto a clean Si substrate with 90 nm SiO$_2$. The PET and samples are heated to 60 °C during the pick-up and to 130 °C for the stamping process. Finally, we dissolve the PET in dichloromethane overnight at room temperature. Contacts (~75 nm gold with a ~5-nm-thick chromium adhesion layer) to the few-layer graphene flakes are made using electron-beam lithography and electron-beam evaporation. Finally, we measure the polarization-dependent SHG on the monolayer TMDs in the heterostructure to determine the twist angle (see Extended Data Fig. 4).

Calibration of hBN dielectric constant
We directly calibrated the hBN dielectric constant against the known dielectric constant of SiO$_2$ using a dual-gate TMD device with a graphite top gate (with hBN as the gate dielectric) and a Si back gate (with SiO$_2$ as the gate dielectric). Specifically, we fabricated a dual-gated MoSe$_2$ device with a 45-nm-thick top hBN gate and a 290-nm-thick SiO$_2$/Si back gate. The hBN crystal is from the same batch that was used to fabricate our WSe$_2$/WS$_2$ moiré heterostructure devices. Extended Data Fig. 3a shows the MoSe$_2$ A exciton peak intensity as a function of the top- and back-gate voltages. Extended Data Fig. 3b shows the extracted charge-neutral points for each Si back gate, which correspond to the top gate voltage at which the system becomes zero net charge. The data show a linear behaviour and the slope indicates the relative gate efficiency. The hBN dielectric constant was then obtained using a parallel-plate capacitor model. The hBN thickness was determined by calibrated atomic force microscopy measurements, and the SiO$_2$ thickness was verified by the optical reflection spectrum. We obtain a hBN dielectric constant of 4.2 ± 0.4 using the SiO$_2$ dielectric constant of 3.9.

Determination of the relative twist angle between WSe$_2$ and WS$_2$ layers
The twist angle between the WSe$_2$ and WS$_2$ flakes in device D1 is 0.4° ± 0.3°, as determined via polarization-dependent SHG (Extended Data Fig. 4). The SHG signal is four times larger on the heterostructure than on the monolayer regions, indicating that this device is closer to 0° than 60°.

Determination of moiré density $n_0$
The moiré density $n_0$ corresponds to one hole per moiré unit cell, and it is directly determined by the moiré periodicity through $n_0 = 1/(L_{\text{M}} \sin(n/3))$. Here $L_{\text{M}} = a/(\delta + \delta^*)$ is the moiré superlattice constant, $\delta = (a - a^*)/a = 4\%$ is the lattice mismatch between WSe$_2$ ($a = 0.328$ nm) and WS$_2$ ($a^* = 0.315$ nm) and $\delta^*$ is the twist angle between the two layers. At $\theta$ smaller than ~1°, $L_{\text{M}}$ is mainly determined by the intrinsic lattice constant mismatch between the two layers, so $n_0$ is not sensitive to a small uncertainty in $\theta$. For device D1, we measured $\theta$ to be 0.4° ± 0.3° using angle-dependent SHG. This corresponds to $n_0 = 1.88 \times 10^{12} \text{ cm}^2$ with an experimental uncertainty of ~10%.

ODRC results from additional near-aligned heterostructures
We measured three near-aligned WSe$_2$/WS$_2$ moiré heterostructures (twist angle <1°). Extended Data Fig. 5a, b shows the ODRC signal of the other two devices, D2 and D3. The qualitative behaviour of the Mott insulator and generalized Wigner crystal states that are observed in D1 and described in the main text is reproducible in these devices. We observe a clear increase in the resistance and decrease in the $\Delta$OC signal at the Mott states and the generalized Wigner crystal state at $n = 1/3n_0$. However, at $n = 2/3n_0$, the generalized Wigner crystal state is almost not observable. We do notice that devices D2 and D3 have much larger inhomogeneous broadening compared with device D1, as shown by the much broader width of the resistance peak. Presumably, the $n = 2/3n_0$...
ODRC signal in a large-twist-angle WSe₂/WS₂ heterostructure

We measured the ODRC signal for a large-twist-angle WSe₂/WS₂ heterostructure, D₄. In this device, the monolayer WSe₂ and WS₂ flakes are intentionally misaligned, and the absorption spectrum is characteristic of a large-twist-angle heterostructure. The signal from the misaligned heterostructure also shows a sharp increase when doped below the bandgap (red curve in Extended Data Fig. 6a), indicating that charge redistribution occurs. However, the signal is largely flat and does not show any clear dips corresponding to insulating states, in sharp contrast with the aligned case (blue curve). This observation is consistent with our conclusion that the insulating states in the aligned heterostructure are Mott and generalized Wigner states in the moiré superlattice, which is not present in a large-twist-angle heterostructure. Extended Data Fig. 6b presents the ODRC signal at several representative frequencies, which shows a characteristic RC circuit fall-off with increasing frequency. No additional feature is observed in the hole-doping region up to a frequency of 1 MHz, further confirming the absence of insulating states. The overall lower resistance in the misaligned device may be due to the different back-gate doping used in the two measurements.

Generation of optical pump–probe pulses with controlled time delay

Two electronic pulse generators (HP 8082A and HP 214B) were used to generate optical pump and probe pulses separately. Both pulse generators were triggered by the digital output of a data acquisition card, so the period and delay of the two triggering signals could be directly controlled with a computer. The output electronic pulses with ~20 ns pulse duration were then converted to optical pulses by two radiofrequency-coupled laser diode modules with energies of 1.80 eV (pump) and 1.66 eV (probe). The pump and probe beams were focused on the sample with beam-spot diameters of ~30 μm and ~5 μm, respectively. Their polarizations were set using linear polarizers and a shared quarter-wave plate. The reflected probe light was collected by a photomultiplier tube. The pump–probe signal was analysed using a lock-in amplifier with modulation frequency of ~2.5 kHz.

Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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Author contributions

F.W. conceived the research. E.C.R., D.W. and C.J. carried out optical measurements. D.W., E.C.R., C.J. and F.W. performed data analysis. E.C.R., D.W., B.G., X.W., M.I.B.U., S.Z., W.Z., Z.Z., J.D.C., M.C. and A.Z. contributed to the fabrication of van der Waals heterostructures. K.Y., M.B. and S.T. grew WSe₂ and WS₂ crystals. K.W. and T.T. grew hBN crystals. All authors discussed the results and wrote the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

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Extended Data Fig. 1 | WSe₂ A exciton gate behaviour. a, Reflection contrast spectra for the lowest-energy WSe₂ A exciton resonance in region 2 of device D1 when the local top-gate voltage $V_{\text{top}}$ is tuned from 0.5 V to −3.5 V. Region 2 is not affected when the hole concentration is tuned in region 1 by $V_{\text{top}}$. b, Reflection contrast spectra for the WSe₂ A exciton in region 2 when the global back gate is tuned from −0.95 V to −1.05 V. The inset shows a zoomed-in view of the exciton peak. The spectral change is monotonic and approximately linear with carrier concentration.
Extended Data Fig. 2 | ODRC signal measured at very low frequencies for a range of modulation voltages. $V_{\text{mod}}$, modulation voltage.
Extended Data Fig. 3 | Calibration of hBN dielectric constant. 

**a**, MoSe₂ A exciton peak intensity measured while tuning the voltages of the top graphite gate and back Si gate. **b**, The extracted charge-neutral points (CNP, dots) for each Si back gate, corresponding to the top graphite gate voltages that bring the system to zero net charge. The black line is a linear fit to the data, from which the relative gate efficiency is determined.
Extended Data Fig. 4 | Determination of WSe₂ and WS₂ flake alignment.
Polarization-dependent SHG signal on monolayer WSe₂ (red circles) and WS₂ (black circles) regions of device D1 and corresponding fittings (red and black curves, respectively).
Extended Data Fig. 5 | ODRC signal for other aligned WSe$_2$/WS$_2$ heterostructures. 

**a, b.** ODRC signal at low (grey) and high (black) frequency from charge-neutral to moderate hole doping in devices D2 (a) and D3 (b). The dashed lines are guides to the eye at hole concentrations of $n = 0$ (purple), $n = n_0/3$ (orange), $n = 2n_0/3$ (green) and $n = n_0$ (blue).
Extended Data Fig. 6 | ODRC signal for a large-twist-angle WSe₂/WS₂ heterostructure. a, Normalized ΔOC for a large-twist-angle heterostructure (D4, blue) and an aligned heterostructure (D1, black). The misaligned heterostructure does not show any insulating features. b, The frequency dependence of the large-twist-angle heterostructure signal shows a characteristic RC circuit fall-off with increasing frequency.