Structural and Optical Properties of Fe$_2$O$_3$ - NiO mixed Thin Films Prepared by Chemical Spray Pyrolysis

Noora J. Mohammed, Nadir F. Habubi$^1$,*
Department of Physics, College of Education, University of Al-Mustansiriya, Baghdad, Iraq

*E-mail address: nadirfadhil@yahoo.com

ABSTRACT

Mixed Fe$_2$O$_3$ - NiO thin films have been prepared by chemical Spray pyrolysis technique onto glass substrate preheated to 450 °C. The thickness of thin films was found to be (400 ±20) nm. XRD measurements reveal that all the film exhibit polycrystalline hexagonal wurztite with a preferred orientation along (104) plane for pure and 10 % and 20 % Ni, while for the rest the structure was amorphous. AFM analysis indicate a nanostructure for all the films. The optical energy gap was found to be increased from optical transitions seems to be direct and the optical energy gap seems to be increased from 2.5 to 2.69 eV as the percentage of NiO increase.

Keywords: Fe$_2$O$_3$; Spray pyrolysis technique; Structural; Optical properties

1. INTRODUCTION

Fe$_2$O$_3$ is one of the most important transition metal with a band gap of 2.2 eV. It received an extensive attention due to its good intrinsic physical and chemical properties, such as its low cost, stability under ambient conditions, and environmentally friendly properties [1]. (α-Fe$_2$O$_3$) is the most stable iron oxide compound material and is widely used in photoelectrodes, gas sensing, catalysts, magnetic recording, and medical fields [2]. Due to its great sensitivity for flammable gases, its fast speed of response and its long term stabilities; Photo electrochemical solar cell, due to its optical band gap, its high optical absorption coefficient; Negative electrode in rechargeable batteries. It is also used for water electrolysis in the presence of sunlight [3,4]. Nickel oxide is a transition metal oxide semiconductor, usually taken as a model for p-type material. NiO is having a wide band gap of 3.6 to 4.0 eV [5] and exhibit rhombohedral or cubic structure, but the most prominent structure was cubic structure [6]. NiO thin films have been studied for applications in electrochromic devices [7], electrode material for Li-ion batteries [8]. Recent works have shown that NiO is also a promising functional material for applications in resistive type gas sensors implementing thin NiO films [9].

Most attractive features of NiO are: (i) excellent durability and electrochemical stability, (ii) low materials cost, (iii) promising ion storage material in terms of cyclic stability, Fe$_2$O$_3$ and NiO films can be prepared by physical and chemical methods such as:
spray pyrolysis [10], electron beam evaporation [11], pulsed laser deposition [12], plasma enhanced chemical vapor deposition [13] and reactive sputtering [14]. It is well known that the structural, optical, surface morphology and electrical properties of materials in thin film form depends on the deposition. This paper represents the fabrication of the mixed oxide Fe$_2$O$_3$-NiO thin film utilizing chemical spray pyrolysis in order to study their structure and optical properties.

2. EXPERIMENTAL

Thin films of iron oxide have been prepared by chemical pyrolysis technique. The spray pyrolysis was done with a laboratory designed glass atomizer, which has an output nozzle about 1 mm. The films were deposited on preheated glass substrates at a temperature of (450 °C). Best homogeneous films were obtained at this temperature. The chemical solution was achieved by adding (1.6221 g) of (FeCl$_3$·6H$_2$O) a source of Fe on 100 ml of deionized water and was (2.377 g) of (NiCl$_2$·6H$_2$O) a source of Ni on 100 ml of deionized water, homogeneous mixture of aqueous solution was achieved by using a magnetic stirrer. The optimized conditions were arrived at the following parameters, spraying time (5 sec), lasted by 2 minutes to avoid excessive cooling, deposition rate (8 ml/min), distance between nozzle and the substrate was (30 ±1 cm) and the carrier gas (filtered compressed air) was maintained at a pressure of 5 bar. Thickness of the samples was measured using the StellarNet TF method and was found to be around 400 ±20 nm. Optical transmittance and absorbance were recorded in the wavelength range (300-900 nm) using UV-Visible spectrophotometer (Shimadzu Company Japan).

3. RESULTS AND DISCUSSION

3.1. Structural properties

Fig. 1 shows the XRD pattern for the as deposited thin films. (XRD) analyses show that (Pure, 10, 20 %) were polycrystalline in nature with a Rhombohedral wurtzite structure and a preferred orientation along (104) plane. A decrease in the values of the crystallization was noticed by the increase ratio of (NiO). All of the observed peaks can be assigned to Fe$_2$O$_3$, in accordance with data from (The Joint Committee on Powder Diffraction Standard) (JCPDS) (0664-33 JCPDS no.) The ratios (30, 40, 50 %) NiO were Amorphous. The Grain size was calculated using the Scherrer equation, and was found to be decreased with increasing (NiO) as shown in Figure (2).

Fig. 3 shows the AFM micrograph for all the films under investigation. The atomic force microscope (AFM) images show that the average roughness and the square root of the average roughness square (RMS) of the films decreased with increasing (NiO) concentrations such as (Pure, 10, 20 %), and increases in the ratios (30, 40, 50 %) NiO. The average Grain size decreased with increasing (NiO) in pure and the ratios (10, 20 %) NiO while increases in (30, 40, 50 %) NiO as shown in the Table 1.
Fig. 1. XRD patterns of films Fe$_2$O$_3$ and Fe$_2$O$_3$ - NiO.

Fig. 2. Grain size for films Fe$_2$O$_3$ and Fe$_2$O$_3$ - NiO.
Fig. 3. Three dimensional atomic force images of films Fe$_2$O$_3$ - NiO.
Fig. 3 (continue). Three dimensional atomic force images of films $\text{Fe}_2\text{O}_3$ - NiO.

$\text{NiO}_{0.1}(\text{Fe}_2\text{O}_3)_{0.9}$

$\text{NiO}_{0.3}(\text{Fe}_2\text{O}_3)_{0.7}$
Fig. 3(continue). Three dimensional atomic force images of films $\text{Fe}_2\text{O}_3$- NiO.

NiO$_{0.4}(\text{Fe}_2\text{O}_3)_{0.6}$

NiO$_{0.5}(\text{Fe}_2\text{O}_3)_{0.5}$
Table 1. Average grain size, surface roughness and RMS values for the as deposited thin films.

| Sample                  | Avg. Diameter (nm) | Surface Roughness (nm) | RMS (nm) |
|-------------------------|--------------------|------------------------|----------|
| Fe₂O₃ (Pure)            | 92.83              | 0.832                  | 1.03     |
| Fe₂O₃ 90 % - NiO 10 %   | 90.49              | 0.772                  | 0.949    |
| Fe₂O₃ 80 % - NiO 20 %   | 82.58              | 0.756                  | 0.944    |
| Fe₂O₃ 70 % - NiO 30 %   | 82.10              | 0.293                  | 0.368    |
| Fe₂O₃ 60 % - NiO 40 %   | 82.42              | 1.11                   | 1.4      |

3.2. Optical properties

Figure (4) shows the optical transmittance spectra of Fe₂O₃ - NiO thin films in the wavelength range of (300-900) nm. The films are transparent in the visible and infrared region of the electromagnetic spectrum the transmittance increased as the NiO content increase The increase in optical transmittance can be attributed to the increase of structural homogeneity and crystallinity [15].

Fig. 4. Variation of (T) as a function of wavelength for Fe₂O₃ - NiO films.
Fig. 4(continue). Variation of $T$ as a function of wavelength for Fe$_2$O$_3$ - NiO films.
Fig. 4(continue). Variation of (T) as a function of wavelength for Fe$_2$O$_3$ - NiO films.
Fig. 4(continue). Variation of (T) as a function of wavelength for Fe$_2$O$_3$ - NiO films.

Figure (5) shows the absorbance of Fe$_2$O$_3$ - NiO films. It is clear that the films have high absorption at short wavelength, the figure show that the absorption decreases with increasing ratios of Nickel.

The change of reflectance as a function of wavelength in the range (300-900) nm is shown in Fig (6) It is seen from the figure that the reflectance decreases with the increased of nickel ratio. The optical band gap of the films is determined by applying Tauc model [16] in the high absorption region by the relation [16]

$$\alpha h\nu = B_o (h\nu - E_g)^r$$  \hspace{1cm} (1)

where $h\nu$ is the photon energy, $E_g$ is the optical energy gap, $B_o$ is a constant. For direct transition $r = 1/2$ or 2/3 and the former value was found to be more suitable for Fe$_2$O$_3$ thin films.

The optical energy gap is determined by plotting $(\alpha h\nu)^2$ as a function of photon energy as shown in Fig. 7.

The value of the optical energy gap were increase with the increasing of nickel content. it is well known that the energy gap of the semiconductor increases with the decrease of the grain.
Fig. 5. Absorbance versus wavelength for Fe$_2$O$_3$ - NiO thin films.
Fig. 5(continue). Absorbance versus wavelength for Fe$_2$O$_3$ - NiO thin films.
Fig. 5(continue). Absorbance versus wavelength for Fe$_2$O$_3$ - NiO thin films.
Fig. 6. The reflectance versus wavelength for Fe$_2$O$_3$ - NiO thin films.

Fig. 7. Variation of $(\alpha h\nu)^2$ vs. photon energy for Fe$_2$O$_3$ - NiO thin film. $E_g = 2.5$ eV.
Fig. 7(continue). Variation of vs. photon energy for Fe$_2$O$_3$ - NiO thin film.

(Fe$_2$O$_3$ 90%) (NiO 10%)

$E_g = 2.52$ eV

$(\alpha h\nu)^2$ (eV/cm)$^2$

Photon Energy $h\nu$ (eV)

(Fe$_2$O$_3$ 80%) (NiO 20%)

$E_g = 2.57$ eV

$(\alpha h\nu)^2$ (eV/cm)$^2$

Photon Energy $h\nu$ (eV)
Fig. 7(continue). Variation of vs. photon energy for Fe$_2$O$_3$ - NiO thin film.

- (Fe$_2$O$_3$ 70%) (NiO 30%)
  - $E_g = 2.6$ eV

- (Fe$_2$O$_3$ 60%) (NiO 40%)
  - $E_g = 2.65$ eV
Fig. 7(continue). Variation of $\varepsilon$ vs. photon energy for Fe$_2$O$_3$ - NiO thin film.

Figure 8 illustrates the variation of $(\varepsilon_r)$ as a function of photon energy. It is clear from the figure that $(\varepsilon_r)$ decrease with increasing of (NiO) content. Figure (9) illustrates

Fig. 8. Variation of $\varepsilon_r$ as a function of wavelength for Fe$_2$O$_3$ - NiO films.
the variation of \( (\varepsilon_i) \) as a function of wavelength. The figure shows a decrease of \( (\varepsilon_i) \) with increasing of wavelength and that, \( (\varepsilon_i) \) decrease with increasing ratio of \( \text{(NiO)} \).

**Fig. 9.** Variation of \( \varepsilon_i \) as a function of wavelength for \( \text{Fe}_2\text{O}_3 \) - NiO films.

**Fig. 10.** Refractive index as a function of wavelength for \( \text{Fe}_2\text{O}_3 \) - NiO thin films.
The refractive indices (n) of the thin films were determined from equation (2) [18-32]. As shown in Fig. (10), the refractive indices of the films are influenced by the NiO content. The refractive indices decrease as the NiO content increases and the refractive index decrease as the wavelength increases.

\[ n_0 = \left( \frac{1+R}{1-R} \right) + \sqrt{\frac{4R}{(1-R)^2} - k_0^2} \] ............(2)

where (n) is the refractive index , (R) is the reflectance.

4. CONCLUSIONS

Fe₂O₃ thin films were prepared using a spray pyrolysis technique onto a glass substrate at temperature of 450 °C. X-ray diffraction (XRD) analyses show that Fe₂O₃ and the ratio of (10, 20 %) were polycrystalline in nature with a Rhombohedral wurtzite structure and a preferred orientation along (104) plane. A decrease in the values of the crystallization was noticed by the increase ratio of (NiO).

The rest ratios (30, 40, 50 %) NiO have now been amorphous.

The atomic force microscope (AFM) images show that the average roughness and the square root of the average roughness square (RMS) of the films decrease with increasing (NiO) concentrations (Pure, 10, 20 %) and increases in the ratios of (30, 40, 50 %) NiO.

From UV-Visible spectrometer, the absorption spectrum was measured also, the optical constant such as real and imaginary dielectric constant were determined and indicated that they decreased with the increase of the wavelength and the of ratio of (NiO) and the value of the optical energy gap increase with increasing ratio of NiO.

References

[1] Wu J. J., Lee Y. L., Chiang H. H., Wong D. K. P., J. Phys. Chem. B110 (2006) 18108.
[2] M. Catti, G. Valerio, Phys. Rev. B 51 (1995) 7441, J. Chen, L. Xu, W. Li, X. Gou, Adv. Mater. 17 (2005) 582.
[3] J. H. Kennedy, D. J. Dunnwald, Electrochem. Soc. 130 (1983) 2013.
[4] B. M. Warnes, F. F. Aplan, G. Simkovich, Solid State Ionics 12 (1984) 271.
[5] H. J. M. Swagten, G. J. Strijkers, P. J. H. Bloemen, M. M. H. Willekens, W. J. M. De Jonge, Phys. Rev. B 53 (1996) 9108.
[6] K. K. Purushothaman, G. Muralidharan, Sol. Energ. Mat. Sol. C. 93 (2009) 1195.
[7] Zhang Xuping, Chen Guoping, Thin Solid Films 298 (1997) 53.
[8] Ying Wang, Ya-Fei Zhang, Hai-Rong Liu, Shen-Jiang Yu, Qi-Zong Qin, Electrochim. Acta 48 (2003) 4253.
[9] I. Hotovy, J. Huran, P. Siciliano, S. Capone, L. Spiess, V. Rehacek, Sensor. Actuat. B-Chem. 103 (2004) 300.
[10] J. D. Desai, Sun-Ki Min, Kwang-Deog Jung, Oh-Shim Joo, *Appl. Surf. Sci.* 253 (2006) 1781.

[11] A. Agrawal, H. R. Habibi, R. K. Agrawal, J. P. Cronin, D. M. Roberts, C. P. R’Sue, C. M. Lampert, *Thin Solid Films* 221 (1992) 239.

[12] M. Tanaka, M. Mukai, Y. Fujimori, M. Kondoh, Y. Tasaka, H. Baba, S. Usami, *Thin Solid Films* 281 (1996) 453.

[13] W. C. Yeh, M. Matsumura, *Jpn. J. Appl. Phys.* 36 (1997) 6884.

[14] M. Bönger, A. Fuchs, K. Scharnagl, R. Winter, T. Doll, I. Eisele, *Sensor. Actuat. B-Chem.* 47 (1998) 145.

[15] Al Asmar R., Zaouk D., Bahouth Ph., Podleki J., Foucaran A., *Microelectronic Engineering* 83 (2006) 393e8.

[16] C. Klingshirn "Semiconductor optics" Second Edition, New York 2005.

[17] C. Shi, G. Wang, N. Zhao, X. Du, J. Li, *Chemical Physics Letters* 454 (2008) 75.

[18] N. Benramdane, W. A. Murad, R. H. Misher, M. Ziane, Z. Kebbab, *Materials Chemistry and Physics* 48 (1997) 119.

[19] N.-T. H. Kim-Ngan, G. Jagło, D. Sitko, W. Soszka, *International Letters of Chemistry, Physics and Astronomy* 2 (2013) 1-10.

[20] Nadir Fadhil Habubi, Sami Salmann Chiad, Saad Farhan Oboudi, Ziad Abdulahad Toma, *International Letters of Chemistry, Physics and Astronomy* 4 (2013) 1-8.

[21] Saad F. Oboudi, Nadir F. Habubi, Ghuson H. Mohamed, Sami S. Chiad, *International Letters of Chemistry, Physics and Astronomy* 8(1) (2013) 78-86.

[22] Sujan Kumar Das, Jahid M. M. Islam, Monirul Hasan, Humayun Kabir, Md. Abdul Gafur, Enamul Hoque, Mubarak A. Khan, *International Letters of Chemistry, Physics and Astronomy* 10(1) (2013) 90-101.

[23] J. A. Najim, J. M. Rozaiq, *International Letters of Chemistry, Physics and Astronomy* 10(2) (2013) 137-150.

[24] Majid H. Hassouni, Khudheir A. Mishjil, Sami S. Chiad, Nadir F. Habubi, *International Letters of Chemistry, Physics and Astronomy* 11 (2013) 26-37.

[25] Harish Kumar, Renu Rani, *International Letters of Chemistry, Physics and Astronomy* 14 (2013) 26-36.

[26] K. K. Patankar, *International Letters of Chemistry, Physics and Astronomy* 1 (2014) 1-8.

[27] Raid A. Ismail, Nadir F. Habubi, Hussam R. Abid, *International Letters of Chemistry, Physics and Astronomy* 4 (2014) 37-47.

[28] Ali M. Mous, Selma M. Al-Jawad, Suad M. Kadhim Al-Shammari, 6 (2014) 16-25.

[29] Hanan R. A. Ali, *International Letters of Chemistry, Physics and Astronomy* 8 (2014) 47-55.

[30] Saba Jameel Hasan, *International Letters of Chemistry, Physics and Astronomy* 9 (2014) 59-70.
[31] Raghad Y. Mohammed, S. Abduol, Ali M. Mousa, *International Letters of Chemistry, Physics and Astronomy* 10 (2014) 91-104.

[32] Raghad Y. Mohammed, S. Abduol, Ali M. Mousa, *International Letters of Chemistry, Physics and Astronomy* 11(2) (2014) 146-158.

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