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Published in:
Engineering Fracture Mechanics

Link to article, DOI:
10.1016/j.engfracmech.2018.12.016

Publication date:
2019

Document Version
Peer reviewed version

Link back to DTU Orbit

Citation (APA):
Juul, K. J., Martínez Pañeda, E., Nielsen, K. L., & Niordson, C. F. (2019). Steady-state fracture toughness of elastic-plastic solids: Isotropic versus kinematic hardening. Engineering Fracture Mechanics, 207, 254-268. https://doi.org/10.1016/j.engfracmech.2018.12.016
Steady-state fracture toughness of elastic-plastic solids:
Isotropic versus kinematic hardening

K. J. Juul\textsuperscript{a,*}, E. Martínez-Pañeda\textsuperscript{b}, K. L. Nielsen\textsuperscript{a}, C. F. Niordson\textsuperscript{a}

\textsuperscript{a}Department of Mechanical Engineering, Solid Mechanics, Technical University of Denmark, DK-2800 Kgs. Lyngby, Denmark
\textsuperscript{b}Department of Engineering, Cambridge University, CB2 1PZ Cambridge, UK

Abstract

The fracture toughness for a mode I/II crack propagating in a ductile material has been subject to numerous investigations. However, the influence of the material hardening law has received very limited attention, with isotropic hardening being the default choice if cyclic loads are absent. The present work extends the existing studies of monotonic mode I/II steady-state crack propagation with the goal to compare the predictions from an isotropic hardening model with that of a kinematic hardening model. The work is conducted through a purpose-built steady-state framework that directly delivers the steady-state solution. In order to provide a fracture criterion, a cohesive zone model is adopted and embedded at the crack tip in the steady-state framework, while a control algorithm for the far-field, that significantly reduces the number of equilibrium iterations is employed to couple the far-field loading to the correct crack tip opening. Results show that the steady-state fracture toughness (shielding ratio) obtained for a kinematic hardening material is larger than for the corresponding isotropic hardening case. The difference between the isotropic and kinematic model is tied to the non-proportional loading conditions and reverse plasticity. This also explains the vanishing difference in the shielding ratio when considering mode II crack propagation as the non-proportional loading is less pronounced and the reverse plasticity is absent.

Keywords: Steady-state, Isotropic hardening, Kinematic hardening, Active plastic zone, Shielding ratio

*Corresponding author
Email address: krjoju@mek.dtu.dk (K. J. Juul)

Preprint submitted to Elsevier December 16, 2019
1. Introduction

The influence of plastic deformation on fracture toughness has been the motivation of a large number of studies in the literature (see e.g. Varias and Shih [1], Tvergaard and Hutchinson [2], Suo et al. [3], Cleveringa et al. [4], Tvergaard [5], Nielsen et al. [6], Nielsen and Niordson [7], Jiang et al. [8], Juul et al. [9]. The common goal has been to achieve a better understanding of the underlying mechanics that affect the toughness of ductile materials by gaining insight into the role of crack tip plasticity. Factors such as rate-dependency (Landis et al. [10]), work hardening (Tvergaard and Hutchinson [11]), strain gradients (Wei and Hutchinson [12], Martínez-Pañeda and Niordson [13]), dynamic lattice effects (Freund and Hutchinson [14]), material property mismatch (Cao and Evans [15], Tvergaard [16]), or micro structure evolution (Kumar and Curtin [17]) affect the fracture properties and determine the extent of crack propagation.

Except for the recent study of mode I cracks by Martínez-Pañeda and Fleck [18], the majority of the published studies of crack growth under monotonic loading confine their focus to isotropic hardening materials, despite the crucial impact of the plastic material response on the shielding ratio. In fact, plastic deformation and the associated dissipation of energy is known to be the main contributor towards enhancing the fracture resistance beyond crack initiation. The far-field loading drives this process and despite being monotonic at the far boundary the conditions experienced by the material passing by the crack tip are very different. It is well documented that material entering the active plastic zone near a steadily growing mode I crack will either exit into an unloading wake or experience reverse plastic loading close to the new fracture surface. Thus, any Bauschinger effect originating from kinematic hardening must have an important influence. Though reversed plasticity does not take place in mode II crack growth, non-proportional loading for material at a distance from the crack face will be demonstrated in the results of the present study. The present study is further motivated by the fact that kinematic hardening effects are expected to play an increasing role in modern structural materials. Composite, multiphase or refined microstructures influence the work hardening response, enhancing kinematic hardening (Ashby [19], de Formanoir et al. [20]).

The main goal of the present study is to investigate how the choice of
hardening model influences the fracture toughness of a steadily growing crack under monotonic mode I, mode II, and mixed mode I/II far-field conditions. In plate tearing steadily growing cracks are encountered when the crack has propagated multiple plate thicknesses. In fact, steady-state is typically reached after crack growth on the order of seven plate thicknesses (Woelke et al. [21], Andersen et al. [22]). Hence, this state can dominate a significant part of the crack path in shell-like structures such as ships, airplanes, and cars. To focus the effort on the part of the propagation path taking place under steady-state conditions, the framework first proposed by Dean and Hutchinson [23] has been adopted and extended to kinematic hardening plasticity. The material steady-state fracture toughness, composed by the energy going into material separation as well as energy dissipated in the surrounding material, is evaluated by introducing the cohesive traction-separation relation proposed by Tvergaard and Hutchinson [2]. This allows for an analysis of the ratio between the external energy applied to the system and the energy specified for the fracture process (referred to as the shielding ratio). Attention is focused on the change in fracture properties when shifting from isotropic hardening to kinematic hardening. Thus, the stress evolution for material points in the vicinity of the crack tip is of particular interest as any deviation from proportional loading will be treated differently in the two types of hardening models.

Throughout this paper, the two types of material hardening and their differences are studied for various conditions of the near tip fracture process (in terms of cohesive zone parameters). Furthermore, the origin of these differences is traced by mapping out the energy dissipation in the vicinity of the propagating crack. In the present study, the material is assumed to be governed by linear hardening. This is chosen to ensure a constant tangent modulus, thus clearly bringing out the essential differences in predictions for steady-state fracture toughness between the isotropic and kinematic hardening models.

The paper is divided into the following sections: The modified boundary value problem is presented in Section 2, the material model, interface model, the algorithm controlling the far-field loading, and the numerical formulation are presented in Section 3, the results are presented in Section 4, and lastly some concluding remarks are stated in Section 5. Throughout, index notation, including Einstein’s summation convention, is used and a superimposed dot, $\dot{}$, denotes the time derivative.
2. Mixed mode boundary layer problem

The steady-state crack propagation study is carried out for mode I/II loading conditions under the assumption of small-scale yielding. To model the continuously growing crack, the steady-state framework presented by Dean and Hutchinson [23] is coupled with a cohesive zone description of fracture, employing the traction-separation law proposed by Tvergaard and Hutchinson [2]. The problem, commonly known as the modified boundary layer problem, is modeled in a 2D plane strain setting as illustrated in Fig. 1 (the considered material properties are collected in Tab. 1). The domain is constructed large enough such that boundary effects do not affect the solution, and the stress intensity factors, $K_I$ and $K_{II}$, can be employed to characterize the stress-field. The mode I/II loading condition is imposed as a far-field condition according to the elastic solution presented by Williams [24] and dictates that the stress field has the form:

$$\sigma_{ij} = \frac{1}{2\pi r} \left( K_I f_{ij}^I(r, \theta) + K_{II} f_{ij}^{II}(r, \theta) \right), \quad (1)$$

where $r$ and $\theta$ are polar coordinates related to the crack tip position, $f_{ij}(r, \theta)$ are dimensionless mode functions, and $K_I$ and $K_{II}$ are the stress intensity factors representing the mode I and II contributions, respectively. Throughout this work, it is assumed that the crack propagates in a straight line (along the $x_1$-direction in Fig. 1). This assumption, common to other mixed-mode crack propagation analyses (see e.g. Tvergaard [5]), constitutes an approximation under mode II dominated loading conditions.

The steady-state fracture toughness is quantified by the so-called crack tip shielding ratio, $K_{ss}/K_0$, which is the stress intensity factor for steady-state crack growth, $K_{ss}$, normalized by the stress intensity factor for crack initiation, $K_0$. The reference stress intensity factor, $K_0$, is defined as:

$$K_0 = \sqrt{\frac{E \Gamma_0}{1 - \nu^2}} \quad (2)$$

where $E$ is Young’s modulus, $\nu$ is Poisson’s ratio, and $\Gamma_0$ is the fracture energy (work of separation of the cohesive zone model). Moreover, any length quantity in the present study is normalized by the reference plastic zone size,
$R_0;$
\[ R_0 = \frac{1}{3\pi} \left( \frac{K_0}{\sigma_y} \right)^2 \]  
(3)

where $\sigma_y$ is the initial yield stress of the material.

3. Constitutive relations and modeling

3.1. Traction-separation relation

The traction-separation relation employed is adopted from Tvergaard and Hutchinson [11, 2]. Accordingly, the traction energy potential is defined as;
\[ \Phi(\delta_t, \delta_n) = \delta_n \int_0^\lambda \sigma(\lambda')d\lambda' \]  
(4)

where $\sigma(\lambda)$ is the traction shown in Fig. 2 as a function of the non-dimensional measure of separation, $\lambda$. The non-dimensional crack separation is defined as; $\lambda = \sqrt{(\delta_n/\delta_n^c)^2 + (\delta_t/\delta_t^c)^2}$, where $\delta_n$ and $\delta_t$ denote the actual separation in the normal and tangential directions, respectively, and the quantities with superscript $c$ denote the corresponding critical values. Thus, as illustrated in Fig. 2, the bond between two nodes completely breaks at $\lambda = 1$. From the traction energy potential, the normal and tangential tractions in the fracture process zone are given as;
\[ T_n = \frac{\partial \Phi}{\partial \delta_n} = \frac{\sigma(\lambda)}{\lambda} \frac{\delta_n}{\delta_n^c} \]  
and  
\[ T_t = \frac{\partial \Phi}{\partial \delta_t} = \frac{\sigma(\lambda)}{\lambda} \frac{\delta_t}{\delta_t^c} \frac{\delta_t}{\delta_t^c}. \]  
(5)

Finally, the work of separation per unit area of interface (the fracture energy) is defined as;
\[ \Gamma_0 = \frac{1}{2} \hat{\sigma} \delta_n^c(1 - \lambda_1 + \lambda_2) \]  
(6)

where $\hat{\sigma}$ denotes the peak traction (cohesive strength) shown in Fig. 2.

3.2. Constitutive models

The framework relies on an infinitesimal strain formulation (both for the isotropic and kinematic model). The small strain formulation has been cho-
sen because previous finite strains studies have shown that the crack propagates at relatively small deformations (Tvergaard and Hutchinson [11]) for the selected values of the cohesive strength. This is also seen in the work by e.g., Wei and Hutchinson [12], Martínez-Pañeda and Fleck [18] where finite strain results are precisely reproduce by a small strain framework. In the infinitesimal strain formulation, the total strains, $\varepsilon_{ij}$, are determined from the displacement gradients; $\varepsilon_{ij} = (u_{i,j} + u_{j,i})/2$. The total strains consist of an elastic part, $\varepsilon_{ij}^e$, and plastic part, $\varepsilon_{ij}^p$, which for an additive split gives the following relationship; $\varepsilon_{ij} = \varepsilon_{ij}^e + \varepsilon_{ij}^p$. Subsequently, the stress field in the rate-independent model is determined from the tensor of instantaneous moduli, $L_{ijkl}$, and the total strain as;

$$\dot{\sigma}_{ij} = L_{ijkl}\dot{\varepsilon}_{kl}. \quad (7)$$

Throughout this work, the material behavior is assumed to be governed by linear hardening such that the tangent modulus, $E_t$, remains constant and given as a fraction of Young’s modulus, $E$. The tangent modulus, $E_t$, enters the instantaneous moduli, $L_{ijkl}$, in Eq. (21).

### 3.2.1. Isotropic hardening

The isotropic model does not consider the Bauschinger effect, as the yield surface expands isotropically in all directions (see Fig. 3a) while maintaining its origin in stress space. The von Mises yield criterion, employed in the present study, takes the form

$$F(\sigma_{ij}) = \frac{3}{2}s_{ij}s_{ij} - (\sigma_e)^2_{\text{max}} = 0, \quad (8)$$

where $s_{ij}$ is the deviatoric stress and $\sigma_e$ is the von Mises stress. In the context of an incremental formulation, the active plastic zone can be evaluated by integrating the total stress in time (Eq. (7)), followed by an evaluation of the criterion for plasticity;

$$\beta = \begin{cases} 1, & \text{for } \sigma_e = (\sigma_e)_{\text{max}} \text{ and } \dot{\sigma}_e \geq 0 \\ 0, & \text{for } \sigma_e < (\sigma_e)_{\text{max}} \text{ or } \dot{\sigma}_e < 0 \end{cases}$$

where $\beta = 1$ indicates a material point governed by plastic loading and $\beta = 0$ indicates a material point governed by elastic unloading (applied in Eq. (21)).
3.2.2. Kinematic hardening

For kinematic hardening, where the yield surface translates in stress space (see Fig. 3b), the yield condition reads:

\[ F(\sigma_{ij}, \alpha_{ij}) = \frac{3}{2} \tilde{s}_{ij} \tilde{s}_{ij} - (\sigma_y)^2 = 0, \]  

(9)

where symbols with (\tilde{\cdot}) denote stress quantities related to the local origin of the translating yield surface, representing the well-known Bauschinger effect. The origin of the translating yield surface, tracked through the back stress, \( \alpha_{ij} \), is used to establish the local stress:

\[ \tilde{\sigma}_{ij} = \sigma_{ij} - \alpha_{ij} \]  

in the yield function, Eq. (9), through the deviatoric stress, \( \tilde{s}_{ij} \). The translation of the yield surface is modeled through an evolution law. Several evolution laws exist in the literature, ranging from simple models to very sophisticated models including effects such as ratchetting and shakedown (Lemaitre and Chaboche [25]). For fracture problems under monotonic loading, reversed loading is expected in the wake of the leading active plastic zone, but repeated cyclic loading is not taking place which means that ratchetting and shakedown are not relevant to the present study. Consequently, the well-known evolution law by Ziegler [26] is chosen, as it includes all the necessary features. Thus, the back stress evolves as:

\[ \dot{\alpha}_{ij} = (\sigma_{ij} - \alpha_{ij}) \dot{\mu} \]  

(11)

where \( \dot{\mu} \) is a proportionality coefficient which is defined as:

\[ \dot{\mu} = \frac{3}{2} \frac{\dot{s}_{ij} \tilde{s}_{ij}}{\sigma_y^2} \]  

(12)

in the present study. For kinematic hardening, where the yield surface translates but maintain its size, the criterion for plasticity is therefore evaluated according to:

\[ \beta = \begin{cases} 
1, & \text{for } \tilde{\sigma}_{e} = \sigma_y \text{ and } \tilde{s}_{kl} \tilde{\sigma}_{kl} \geq 0 \\
0, & \text{for } \tilde{\sigma}_{e} < \sigma_y \text{ or } \tilde{s}_{kl} \tilde{\sigma}_{kl} < 0 
\end{cases} \]
3.3. Steady-state framework

The fracture toughness at steady-state, $K_{ss}$, can be approximated with traditional incremental numerical methods by computing the crack growth resistance curve. However, such methods often suffer from convergence issues and are inefficient as they are forced through the transient regime of a problem before reaching the steady-state. To avoid such issues, a steady-state framework is employed, leading to accurate predictions of the steady-state fracture toughness at a fraction of the computational cost. The steady-state framework presented builds upon an extension of the procedure suggested by Dean and Hutchinson [23] to account for a kinematic hardening law.

The pivotal step is to utilize the nature of a steady-state problem to determine the history dependent field quantities. The steady-state condition is noticed for an observer located at the tip of a continuously growing crack when the field quantities that surrounds the crack tip are no longer subject to changes. Any time derived quantity, $\dot{f}$, in the constitutive equations are then transformed into spatial derivatives, through the crack propagation speed, $\dot{a}$, in the direction of the material flow (negative $x_1$-direction, see Fig. 1) according to the relation:

$$\dot{f} = -\dot{a} \frac{\partial f}{\partial x_1}.$$ (13)

Thus, any total quantity at a given material point $(x_1^*, x_2^*)$, is evaluated through spatial integration, starting upstream in the undeformed elastic material ahead of the crack tip $(x_1^0, x_2^*)$, and following a streamline (material flow line) until it reaches the point of interest $(x_1^*, x_2^*)$ downstream (see e.g. Juul et al. [9, 27]). Thus, the loading history at a given material point $(x_1^*, x_2^*)$ is retrieved from all the upstream points along the streamline, representing earlier states.

The steady-state framework is based on the conventional principle of virtual work (PWV) for quasi-static problems:

$$\int_V \mathcal{L}_{ijkl} \varepsilon_{kl} \delta \varepsilon_{ij} dV + \int_{S_c} T_i \delta u_i dS_c = \int_S t_i \delta u_i dS + \int_V \mathcal{L}_{ijkl} \varepsilon_{ij} \delta \varepsilon_{kl} dV$$ (14)

where $t_i = \sigma_{ij} n_j$ is the surface traction, $T_i$ is the traction from the traction-separation law, and $\mathcal{L}_{ijkl}$ is the isotropic elastic stiffness tensor. The volume analyzed is denoted $V$, $S_c$ is the interface (cohesive) surface, and $S$ is the bounding surface, with $n_j$ denoting the unit outward normal vector.

The algorithm employed for the spatial integration procedure is outlined
below, where it is seen that the main difference is in the integration of the back stress in step 4 ($m$ refers to the iteration number):

1. Use the plastic strains, $\varepsilon_{ij}^{(m-1)}$, to determine the current displacement field, $u_{i}^{(m)}$ ($\varepsilon_{ij}^{(m)}$ is assumed to be zero in the first iteration).

2. Determine the total strain, $\varepsilon_{ij}^{(m)}$, from the displacement field, $u_{i}^{(m)}$.

3. Determine the total stress field outside the streamline domain:
   - Stresses can be determined directly from the total strain ($\varepsilon_{ij}^{(m)} = 0$):
     \[ \sigma_{ij}^{(m)} = \mathcal{L}_{ijkl} \varepsilon_{kl}^{(m)} \] (15)
     where $\mathcal{L}_{ijkl}$ is the elastic stiffness tensor.

4. Determine the stresses inside the streamline domain:
   - Determine the spatial derivative of the stress:
     \[ \frac{\partial \sigma_{ij}^{(m)}}{\partial x} = L_{ijkl}^{(m)} \frac{\partial \varepsilon_{kl}^{(m)}}{\partial x_{1}} \] (16)
     where $L_{ijkl}$ is the tensor of instantaneous moduli.
   - Determine the spatial derivative of the back stress (only for kinematic hardening):
     \[ \frac{\partial \alpha_{ij}^{(m)}}{\partial x_{1}} = (\sigma_{ij} - \alpha_{ij}) \frac{\partial \mu_{ij}^{(m)}}{\partial x_{1}} \quad \text{with} \quad \frac{\partial \mu_{ij}^{(m)}}{\partial x_{1}} = \frac{3}{2} \frac{s_{ij}}{\sigma_{y}^{2}} \frac{\partial \sigma_{ij}^{(m)}}{\partial x_{1}} \] (17)
   - Perform spatial integration along streamlines:
     \[ \sigma_{ij}^{(m)} = \int_{x_{1}^{0}}^{x_{1}} \frac{\partial \sigma_{ij}^{(m)}}{\partial x_{1}} \, dx_{1} \quad \text{and} \quad \alpha_{ij}^{(m)} = \int_{x_{1}^{0}}^{x_{1}} \frac{\partial \alpha_{ij}^{(m)}}{\partial x_{1}} \, dx_{1} \] (18)

5. Determine the plastic strain field, $\varepsilon_{ij}^{(m)} = \varepsilon_{ij}^{(m)} - \mathcal{M}_{ijkl}^{(m)} \sigma_{kl}^{(m)}$, in the streamline domain, with $\mathcal{M}_{ijkl}$ being the elastic compliance tensor.

6. Repeat 1 to 5 until solution is converged.

In the algorithm, the following constitutive tensors have been applied
which include the elastic stiffness tensor,

\[
\mathcal{L}_{ijkl} = \frac{E}{1 + \nu} \left[ \frac{1}{2} (\delta_{ik}\delta_{jl} + \delta_{il}\delta_{jk}) + \frac{\nu}{1 - 2\nu} \delta_{ij}\delta_{kl} \right],
\]

(19)

the elastic compliance tensor,

\[
\mathcal{M}_{ijkl} = \frac{1}{E} \left[ \frac{1 + \nu}{2} (\delta_{ik}\delta_{jl} + \delta_{il}\delta_{jk}) - \nu \delta_{ij}\delta_{kl} \right],
\]

(20)

and the isotropic tensor of instantaneous moduli,

\[
L_{ijkl} = \mathcal{L}_{ijkl} - \beta \frac{3}{2} \frac{E/E_t - 1}{E/E_t - (1 - 2\nu)/3} \frac{s_{ij}s_{kl}}{\sigma_e^2},
\]

(21)

where it should be noted that in the tensor of instantaneous moduli, \( s_{ij}s_{kl}/\sigma_e^2 \) is substituted with \( \tilde{s}_{ij}\tilde{s}_{kl}/\sigma_y^2 \) for the kinematic hardening model.

The numerical stability of the steady-state algorithm is in general better than for incremental frameworks, although, certain problems can arise in areas with steep gradients such as at the crack tip. In order to limit such stability issue of the algorithm a minor change has been made to the original procedure by Dean and Hutchinson [23] following the suggestion by Niordson [28] and Nielsen and Niordson [7], where a sub-increment procedure between Gauss points has been introduced in the spatial integration scheme.

### 3.4. Cohesive elements and traction-separation relation

The implementation of the cohesive zone builds upon the standard case presented by e.g. del Busto et al. [29]. However, for the steady-state framework, minor modifications are introduced when building both the stiffness matrix and the right-hand side of the equation system. When discretizing the virtual work principle in Eq. (14), the nonlinear part of the contribution from the cohesive elements is moved to the right-hand side and acts as a force term. Hence, the discretized system reads:

\[
\left( \int_V [B]^T \mathcal{L} [B] dV + \int_{S_c} [B_c]^T \frac{\partial \{ T^{\text{ini}} \}}{\partial \{ \delta \}} [B_c] dS_c \right) \{ U \} \\
= \int_S [N]^T \{ t \} dS + \int_V [B]^T \mathcal{L} \{ \varepsilon^p \} dV + \int_V [B_c]^T \{ T^{\text{lin}} \} - \{ T^{\text{act}} \} dS_c
\]

(22)
where $\partial\{T^{\text{ini}}\}/\partial\{\delta\}$ is the initial slope of the traction-separation curve in the region $\lambda < \lambda_1$ (see Fig. 2), $T^{\text{ini}}$ is the corresponding traction predicted if a linear curve with the initial slope is followed, and $T^{\text{act}}$ is the traction obtained from the actual curve in Fig. 2. The standard strain-displacement and shape function matrices, $[B]$, $[B_c]$ and $[N]$, are given in the work by del Busto et al. [29]. The partition of the cohesive contribution is necessary for the steady-state framework because the equations are no longer expressed on incremental form. However, the partition also entails a considerable reduction in the computational cost as the system matrix only needs to be built and factorized once, with subsequent iterations relying solely on back-substitution.

In the present study, the equation system is discretized using quadratic 8-node isoparametric elements evaluated through $2 \times 2$ Gauss points and quadratic 6-node isoparametric cohesive elements evaluated through 8 Gauss points.

3.5. Control algorithm for the boundary layer problem

The far-field required to drive crack propagation in an elastic-plastic solid is generally unknown and, thus, to efficiently study the steady-state fracture toughness (the shielding ratio) it is necessary to implement a scheme to control the applied far-field loading such that the energy needed for steady-state crack propagation is provided. For this purpose, several techniques have been proposed in the literature. In the present study, the idea employed by Segurado and LLorca [30] and Martínez-Pañeda and Fleck [31] is adopted as it offers fast and stable convergence while at the same time lending itself nicely to implementation in the developed numerical scheme. The technique is here generalized to treat problems with mode mixity. The key idea behind the procedure is to link the loading history to a monotonically increasing parameter that is not affected by potential instabilities. For the present investigation of steady-state crack growth such a parameter could be the crack tip opening displacements (normal and tangential separation) as such quantities should be non-decreasing at all times during loading - or, more precisely, kept fixed in the steady-state framework (schemes with other or more parameters included can also be found in Segurado and LLorca [30]).

Once suitable monotonically increasing parameters have been chosen, a connection between these parameters and the load at the outer far boundary must be established. The connection is introduced through; i) two constraint equations (see Eq. (23)) that ensure the prescribed crack tip opening, ii) a
global equilibrium equation which ensures that the loading on the crack tip, enforced by the constraint equations, is balanced out by the loading on the outer far boundary, and iii) a set of geometrical constraints that ensure a smooth distribution of the far-field loading according to the elastic solution by Williams [24]. The two additional constraint equations are introduced by adding two additional degrees of freedom, $Q_n$ and $Q_t$, to the system (with corresponding right-hand sides, $\Delta_n$ and $\Delta_t$). The desired crack tip opening, $\Delta_n$ and $\Delta_t$, is then enforced through the following displacement constraints:

$$u^{(N_1)}_1 - u^{(N_2)}_1 = \Delta_t \quad \text{and} \quad u^{(N_2)}_2 - u^{(N_2)}_1 = \Delta_n,$$

where $N_1$ and $N_2$ refers to the two nodes located at the crack tip on each side of the crack plane (see Fig. 4). In the case of a pure mode I loaded crack, the enforced tip displacements are; $\Delta_n = \delta^c_n$ and $\Delta_t = 0$, while the pure mode II crack is analyzed by enforcing; $\Delta_n = 0$ and $\Delta_t = \delta^c_t$. Here, $\delta^c_n$ and $\delta^c_t$ are the critical normal and tangential separation, respectively, related to the cohesive traction-separation relation (see Section 3.4).

The constraint equations in Eq. (23) introduce reaction forces at the crack tip that drive the opening. This is, however, artificial and does not resemble crack growth under far-field mode I/II loading. To ensure proper far-field loading of the crack tip, a coupling to the outer far boundary is created through two global equilibrium considerations such that the reaction force at the crack tip becomes zero. This is enforced by adding the following contributions to the system matrix, $K$, for a given node $m$:

$$K((N_{BC})_1, m)u^{(m)} + Q_t = 0 \quad \text{and} \quad K((N_{BC})_2, m)u^{(m)} + Q_n = 0$$

where $(N_{BC})_i$ refers to the global degrees of freedom of an arbitrary node, $N_{BC}$, on the outer far boundary (see Fig. 4). The coupling ensures that the chosen node on the far boundary, $N_{BC}$, will displace according to the prescribed crack tip opening, as dictated by equilibrium, i.e. the far-field becomes an outcome of the equilibrium solution. Finally, a set of geometric constraints are defined to ensure that the displacement of all the nodes at the outer boundary is consistent with the displacements of the arbitrary node, $N_{BC}$. The geometric constraint is determined from the elastic far-field solution to the boundary layer problem presented in Section 2. In terms of the displacement field, the elastic far-field of node $m$, at the outer far
boundary, reads:

\[
\begin{align*}
    u_1^{(m)}(r_{(m)}, \theta_{(m)}) &= C(m) \left( K_I f_{K_{1I}}^{u_1^{(m)}}(r_{(m)}, \theta_{(m)}) + K_{II} f_{K_{1II}}^{u_2^{(m)}}(r_{(m)}, \theta_{(m)}) \right) \\
    u_2^{(m)}(r_{(m)}, \theta_{(m)}) &= C(m) \left( K_I f_{K_{1I}}^{u_2^{(m)}}(r_{(m)}, \theta_{(m)}) - K_{II} f_{K_{1II}}^{u_1^{(m)}}(r_{(m)}, \theta_{(m)}) \right)
\end{align*}
\]  

(26)

where \( C_m = 1/(2\mu)\sqrt{r_{(m)}/(2\pi)} \) and \( f \) are displacement mode functions. Thus, the stress intensity factors, \( K_I \) and \( K_{II} \), corresponding to a known displacement, \( u_i^{(m)} \), of a given node, \( m \), on the outer far boundary can be determined by inverting Eq. (26), such that:

\[
\begin{align*}
    K_I &= \frac{f_{K_{1I}}^{u_1^{(m)}} u_1^{(m)} + f_{K_{1II}}^{u_2^{(m)}} u_2^{(m)}}{C(m) \left( f_{K_{1I}}^{u_1^{(m)}} f_{K_{1I}}^{u_1^{(m)}} + f_{K_{1II}}^{u_2^{(m)}} f_{K_{1II}}^{u_1^{(m)}} \right)} \\
    K_{II} &= -\frac{f_{K_{1I}}^{u_2^{(m)}} u_1^{(m)} - f_{K_{1II}}^{u_1^{(m)}} u_2^{(m)}}{C(m) \left( f_{K_{1I}}^{u_1^{(m)}} f_{K_{1I}}^{u_1^{(m)}} + f_{K_{1II}}^{u_2^{(m)}} f_{K_{1II}}^{u_1^{(m)}} \right)}
\end{align*}
\]  

(27)

That is, the \( K \)-field applied to the outer far boundary can be linked to the displacement of the arbitrary node, \( N_{BC} \), and subsequently to the remaining nodes on the outer far boundary. In this way, the multi-point geometric constraint is stated as:

\[
\begin{align*}
    u_1^{(m)} &= C(m) \frac{f_{K_{1I}}^{u_1^{(m)}} f_{K_{1I}}^{u_2^{(N_{BC})}} + f_{K_{1II}}^{u_1^{(m)}} f_{K_{1II}}^{u_2^{(N_{BC})}}}{C(N_{BC}) f_{K_{1I}}^{u_1^{(N_{BC})}} f_{K_{1II}}^{u_1^{(N_{BC})}} + f_{K_{1I}}^{u_2^{(N_{BC})}} f_{K_{1II}}^{u_2^{(N_{BC})}}} u_1^{(N_{BC})} \\
    u_2^{(m)} &= C(m) \frac{f_{K_{1I}}^{u_2^{(m)}} f_{K_{1I}}^{u_2^{(N_{BC})}} - f_{K_{1II}}^{u_1^{(m)}} f_{K_{1II}}^{u_2^{(N_{BC})}}}{C(N_{BC}) f_{K_{1I}}^{u_1^{(N_{BC})}} f_{K_{1II}}^{u_1^{(N_{BC})}} + f_{K_{1I}}^{u_2^{(N_{BC})}} f_{K_{1II}}^{u_2^{(N_{BC})}}} u_2^{(N_{BC})}
\end{align*}
\]  

(28)

The multi-point geometric constraint is enforced directly on the stiffness
matrix. In combination with the constraint equation in Eq. (23) this ensures a distributed loading on the outer far boundary according to the \( K \)-field required to enforce the prescribed crack tip displacements (\( \Delta_n \) and \( \Delta_t \)).

4. Results

Two types of analyses are conducted to gain better insight into how the hardening model affects the fracture toughness at steady-state crack growth in an elastic-plastic solid. First, the shielding ratio, \( K_{ss}/K_0 \), for various fracture process zone conditions (controlled by the cohesive zone) is investigated to quantify the differences between isotropic and kinematic hardening predictions (Section 4.1). Results are obtained for pure mode I, pure mode II and mixed-mode steady-state crack propagation. Secondly, the origin of the notable differences observed in the shielding ratio is investigated through the energy dissipation density in the isotropic and kinematic hardening materials (Section 4.2). The energy dissipation density is investigated in an attempt to identify which regions of the active plastic zone in the vicinity of the crack tip that primarily controls shielding by adding to the steady-state fracture toughness. This investigation is conducted by tracing material points traveling along the crack path (in the \( x_1 \)-direction) at different distances from the crack face.

The study employs a mesh with a total of 310,000 elements in the entire domain, where approximately 60,000 of the elements are located in the region of the main plastic zone. Furthermore, the critical normal and tangential separation will be related to the mesh such that; \( \delta_n^c = \delta_t^c = 0.2L_{e,min} \), where \( L_{e,min} \) is the minimum element length in the domain.

4.1. Crack tip shielding ratio

To quantify the differences between isotropic and kinematic hardening, the shielding ratio, \( K_{ss}/K_0 \), is studied by employing the cohesive zone model presented in Section 3.1. The cohesive zone model ensures that the energy release rate required for crack propagation is identical for both the isotropic and kinematic material while the far-field is scaled accordingly.

The crack tip shielding ratio, \( K_{ss}/K_0 \), is presented as a function of the normalized peak traction, \( \hat{\sigma}/\sigma_y \), in Fig. 5 for a pure mode I crack (\( K_{II} = 0 \)) for both isotropic (dashed lines) and kinematic hardening (solid lines). For low hardening (\( E_t = E/100 \)) it is seen that the shielding ratio is almost identical for the two materials as hardening has limited influence on the predicted
stress level (close to perfectly plastic). That is, the yield surface remains close
to identical for the two types of material hardening. However, a significant
shift is seen for the isotropic model when the strain hardening is increased
to \( E_t = E/20 \), while the effect in the kinematic model is small. Similarly,
by choosing an even higher hardening, \( E_t = E/10 \), an even greater shift is
seen for the isotropic material, whereas the shift for the kinematic material
is much less pronounced. It is important to notice that for fixed strain hard-
ening and peak traction in the cohesive zone, the kinematic material always
predicts either an equal or greater shielding ratio compared to that of the
isotropic material model. Furthermore, it is seen that shield ratio for the
kinematic hardening solid appears to be unbounded for peak tractions larger
than 2.9, 3.2 and 4.4 times the yield stress for \( E_t = E/100 \), \( E_t = E/20 \) and
\( E_t = E/10 \), respectively.

The shielding ratio for a pure mode II loaded crack \((K_I = 0)\) is presented
in Fig. 6. Contrary to the pure mode I crack, a much less significant effect is
observed when changing from an isotropic to a kinematic hardening model.
As expected, the curves are almost identical for the case of \( E_t = E/100 \)
(and essentially coincides in the figure), but the difference in the predicted
shielding ratio remains limited when increasing the strain hardening. When
compared to the mode I study, it is also observed that similar values of the
steady-state fracture toughness are attained for significantly lower cohesive
strengths. The reason for this is that the stress triaxiality close to the crack
tip is significantly smaller in mode II, promoting plastic deformation at lower
load levels, as specified in terms of the stress intensity factor. Furthermore,
as discussed by Tvergaard [5], values in closer agreement between mode I
and mode II would be attained if the crack was allowed to grow along its
preferred path.

Finally, the shielding ratio is studied for the mixed mode case where
\( K_I = K_{II} \). Figure 7 reveals a shielding ratio for mixed mode conditions
which is located between the pure mode I and pure mode II predictions.
However, it should be noted that the contribution from each mode in an
elastic-plastic solid relies on the crack tip conditions and is therefore not
directly reflected by the far-field mode mixity.

4.2. Active plastic zones and energy dissipation

The investigation of steady-state fracture toughness showed a larger shield
ratio for kinematic hardening solids, independently of the loading mode. To
assess the influence on the material behavior in the vicinity of the crack
tip, the plastic zone is presented in Fig. 8 for both isotropic and kinematic hardening for the mode I case with tangent modulus $E_t = E/20$ and peak traction $\hat{\sigma}/\sigma_y = 3.5$. As expected, the plastic zone is significantly larger in the kinematic hardening case to accommodate the larger dissipation.

The different energy dissipation levels in the two hardening models are investigated through the plastic zone shapes and the associated energy dissipation densities by applying identical far-field loading conditions (equal energy input to the system) in the two models. A direct consequence of this study is that the energy at the crack tip depends on the energy dissipation in each model, i.e. the fracture criterion is not identical. For this study, the cohesive elements are removed from the model i.e. the crack plane is now represented by a line of single nodes rather than dual nodes such that crack tip opening does not take place and the crack remains perfectly sharp. The level of hardening remains fixed at $E_t = E/20$. Furthermore, in the study of the plastic zones, any length quantity is normalized by Irwin’s approximation of the plastic zone size $R_p$, identical to Eq. (3), but with the far-field, $K$, instead of the initiation threshold $K_0$ (similarly, $\Gamma$ is computed from the far-field, $K$).

The active plastic zones for pure mode I and mode II loading are presented in Fig. 9. For a mode I crack (see Fig. 9a), the effect of the kinematic model is most pronounced close to the crack tip - especially on the downstream side (left in Fig. 9) of the primary active plastic zone where the material experiences reversed loading. The isotropic model is dominant, in terms of the width of the active plastic zone, further away from the crack face on the upstream side (right in Fig. 9). For the mode II loaded crack (see Fig. 9b), no significant differences between the plastic zone for the isotropic and kinematic hardening models are observed, consistent with the shielding ratio study. This is largely tied to the nature of a mode II loaded crack tip as reversed loading is absent and the non-proportionality is less severe compared to the mode I crack.

To further investigate the differences between the two hardening models, the energy dissipation density, $w$, is extracted along the four horizontal paths illustrated in Fig. 9. On each path, the energy dissipation density is evaluated as:

$$w = - \int_{-\infty}^{\infty} \left( \sigma_{ij} \frac{dx_p^p}{dx_1} \right) dx_1,$$  \hspace{1cm} (29)$$

according to the streamline integration scheme. The energy dissipation den-
sity is presented in Fig. 10 for mode I loading. Here, it is seen that energy
dissipation density continuously grows from the onset of the plastic zone un-
til the steady-state level is reached far behind the crack tip. Common to
all paths is that the kinematic hardening entails a larger energy dissipation
density. The largest difference is seen closest to the crack plane (path 1) in
the unloading region immediately behind the crack tip where the kinematic
model dissipates more energy than the isotropic model. The kinematic model
continues to have the largest dissipation density for path 2 and 3 albeit the
difference becomes progressively smaller when moving away from the crack
plane as the loading becomes less complex and less severe, affecting both
the level of plasticity and the non-proportionality. The trend is, however, in-
verted for path 4 as the isotropic solution here displays the largest dissipation
density. Nevertheless, the magnitude of the energy dissipation density in this
region is comparably low and has little influence on the overall dissipation.

To identify possible sources of the difference in the energy dissipation
density for a mode I loaded crack, the translation of the kinematic yield
surface is investigated. The translation of the kinematic yield surface is
represented by the effective back stress:

\[ \alpha_e = \sqrt{3\alpha_{ij}'\alpha_{ij}'}/2. \] (30)

In Fig. 11 it is seen that for path 1 and 2 (see Fig. 9a), the yield surface con-
tinues to translate in stress space as reversed plasticity takes place behind
the crack tip. Ultimately, the effective back stress profile reaches a short
unloading plateau further behind the crack tip. This continued movement of
the kinematic yield surface causes additional dissipation when compared to
the isotropic model. As non-proportional loading or reversed plasticity takes
place the kinematic hardening solid experiences a greater change to the yield
surface normal (larger curvature compared to the expanded isotropic yield
surface) and hence the plastic strain increment undergoes a larger change in
direction (or magnitude). In fact, this observation is partially supported by
Tvergaard [32], investigating sheet metal necking using a kinematic harden-
ing model. Here, the kinematic hardening was observed to promote localiza-
tion (similar to corner theories, see Mear and Hutchinson [33]) due to the
curvature of the kinematic yield surface. This suggests that non-proportional
loading has a significant impact in a kinematic model. For path 3 and 4, it is
seen that the back stress reaches a constant value directly due to the absence
of reversed plasticity in these regions. This observation, combined with less severe non-proportionality of the loading in this region, supports the smaller difference in dissipation density observed in Fig. 10.

In contrast to the mode I loaded crack, the difference in the dissipation density is negligible when comparing an isotropic and a kinematic hardening material for a mode II loaded crack. Figure 12 shows results along the four paths shown in Fig. 9b. For path 1, the dissipation density is slightly higher for isotropic hardening, whereas for path 2, 3, and 4 the energy dissipation density is slightly higher for the kinematic model. Thus, for the mode II crack, differences are marginal and no distinct region is observed where one model clearly dominates.

The effective back stress along the four paths in Fig. 9b is shown in Fig. 13. Here it is seen that the yield surface translates in the stress space until the crack tip is reached and remains stationary (constant $\alpha_e$) behind the crack. That is, no reversed plasticity is predicted for the mode II loaded crack, effectively eliminating one source of additional energy dissipation in the kinematic hardening model relative to the mode I analysis. In addition, the non-proportional loading is expected to be much less severe for the mode II crack, rationalizing the similar steady-state fracture toughness predictions obtained with isotropic and kinematic hardening models, respectively, (recall Fig. 6).

5. Concluding remarks

A steady-state framework combined with a cohesive zone model has been developed for the purpose of studying the difference in cracks propagating in either isotropic or kinematic hardening materials. The study focuses on the plastic zone size, the evolution of the strain energy dissipation density, and the shielding ratio for mode I, mode II and mode I/II cracks. The main findings are:

- The shielding ratio is generally largest for the kinematic hardening material compared to an isotropic hardening material. The effect is most significant for a mode I crack whereas the effect is very limited for a mode II crack. Under mixed mode loading conditions, the shielding ratio falls between that of the two pure modes.

- The active plastic zone for a mode I crack is slightly larger for a kinematic hardening material in the central half of the active plastic zone,
whereas the isotropic hardening material has a larger plastic zone in
the exterior half of the active plastic zone, for identical far-field loading.
For the mode II crack, there is no significant difference in the plastic
zone shapes.

- The largest difference in energy dissipation density between the isotropic
  and kinematic hardening material for a mode I crack is observed close
to the crack face, behind the crack tip, in the region of reverse loading.
  For the mode II crack, a very small difference in the dissipation density
  is observed. The dissipation density is slightly larger close to the crack
  face for isotropic hardening whereas further away from the crack face
  the kinematic model becomes dominant. The difference is, however,
  very small.

  The main sources of the larger energy dissipation for kinematic harden-
ing is attributed to the stronger path dependence associated with a larger
curvature of the yield surface (sensitivity to non-proportional loading) and
the reverse loading prone to initiate plasticity sooner than for the isotropic
hardening material (smaller yield surface). The large differences observed in
mode I conditions between isotropic and kinematic predictions imply that
fracture toughness estimations from R-curve modeling are very conservative.
This could have important implications in damage tolerant design in the
aerospace or energy sectors, among others.

6. Acknowledgement

The work is financially supported by The Danish Council for Indepen-
dent Research in the project “New Advances in Steady-State Engineering
Techniques”, grant DFF-4184-00122. E. Martínez-Pañeda additionally ac-
knowledges financial support from the Ministry of Economy and Compet-
itiveness of Spain through grant MAT2014-58738-C3 and the People Pro-
gramme (Marie Curie Actions) of the European Unions Seventh Framework
Programme (FP7/2007-2013) under REA grant agreement n 609405 (CO-
FUNDPostdocDTU).
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| Parameter     | Significance          | Value       |
|---------------|-----------------------|-------------|
| $\sigma_y/E$  | Yield strain          | 0.003       |
| $\dot{\sigma}$ | Peak normal traction  | $0.3 - 5.3\sigma_y$ |
| $\nu$         | Poisson’s ratio       | 0.33        |
| $E/E_t$       | Tangent modulus       | 10 – 100    |
| $\lambda_1$  | Shape parameter       | 0.15        |
| $\lambda_2$  | Shape parameter       | 0.5         |

Table 1: Material Properties.