Adhesive Strength of Hexadecane on Different Iron Compounds: an MD Approach

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ABSTRACT
The lubricity of alkane is a research target for numerous tribological applications in either industrial area or fundamental scientific studies. In the current work, a comparative investigation using a classical molecular dynamics (MD) method is carried out to investigate the effect of pure iron and its oxide surfaces on structural properties, adsorption ability of hexadecane (C_{16}H_{34}). A reliable force field (FF) of condensed-phase optimized molecular potentials for atomistic simulation studies (COMPASS) is employed to describe the intra- and intermolecular interactions for hexadecane and its interaction with iron oxide surfaces, while the interaction between hexadecane and pure iron is derived from an ab initio result. Regarding the surfaces, the pure iron surfaces are considered using embedded-atom method/Finnis-Sinclair potential (EAM/FS), while the iron oxide surfaces are constructed using the traditional Buckingham force field. The results reveal that hexadecane shows preferential adsorption on iron oxide surfaces compared to pure iron.

KEYWORDS
Molecular dynamics simulation; Hexadecane; Adsorption; Iron oxides; Iron.

1. Introduction

Hexadecane has been used widely as a base oil lubricant for several decades. Several experimental investigations have been carried out to investigate the adsorption of linear hydrocarbon on metal and metal oxide surfaces. There were also plenty theoretical studies investigating the structural and dynamic properties of alkanes on the solid surfaces [1-3]. In some theoretical works [4-7] and experimental studies [8, 9], the authors indicated that the linear hydrocarbon adsorbed molecularly on metal and metal oxide surfaces. Compared with graphitic carbon, hydrocarbon interacted weakly on MgO(100) surface, but the adhesive strength became larger on Pt(111) [1, 9]. Although this kind of hydrocarbon has attracted the great attention from researchers, the insight into its adhesive strength with steel surface is not accomplished yet. Wetterer et al. explained for the linear relationship between the desorption energies and chain length, the authors disclosed that the flat alignment of linear hydrocarbon was the source of similar additive energy for each methyl group [3]. However, this linear relationship was retained for the chain length within five methyl groups, a nonlinear relationship was found for the larger number of carbon atoms [10]. This observation was confirmed by an MD simulation by Li et al. [5] who studied the adsorption of linear alkanes on α-Al_{2}O_{3}(0001) surface. For pure transition metal surface, such as Fe, Ni, Pt …, the obtained adhesive strength was much larger for short hydrocarbon. For instances, Anderson et al. found that methane reacted on iron surface with an activation energy of roughly 88 kJ/mol [11-14]. The smaller values have been found for other surfaces of Ni and Pt.

This work investigates the adsorption of hexadecane on iron and its oxide surfaces, and analyses the influence of crystalline structure on thin adsorbed film. This article was organised in the following order: (1) the implemented methodology and parameters for current calculation, (2) a parameterization procedure for hexadecane-iron adhesive strength to determine the pairwise interaction between iron and hexadecane atoms, (3) the calculation of surface structure for each type of surface and their adsorption energies with hexadecane.
2. Methodology

The investigation into lubricant adhesive strength, and the surface properties of both solid and fluid were implemented using the FORCITE module in the commercial software Material Studio 7.0 (MS 7.0). COMPASS FF was applied for lubricant [15], while Buckingham FF was used to construct the iron oxide surfaces. The embedded-atom method/Finnis-Sinclair potential (EAM/FS) was used to describe pure iron surfaces [16]. The parameters for Buckingham potential, as reported by Guillot et al., are presented in Table 1. This potential reproduced appropriate thermodynamic, structural, and transport properties of FeO and Fe₂O₃ compounds in natural silicate melt at low and high pressures [17, 18]. The interactions between surface atoms with fluid atoms and between different lubricant molecules were determined by using Lenard-Jones 9-6 (L-J 9-6) potential with a cut-off distance of 12.5 Å combined with a long-range Columbic interaction. The combination rules for this L-J 9-6 potential were given by the equations 1-2.

\[
\begin{align*}
    r_{ij}^6 &= \left( \frac{(r_i^0)^6 + (r_j^0)^6}{z} \right)^{\frac{1}{6}} \\
    \varepsilon_{ij}^0 &= \frac{z}{\varepsilon_i^0 \varepsilon_j^0} \left( \frac{(r_i^0)^6 + (r_j^0)^6}{r_i^0 + r_j^0} \right)^{\frac{3}{2}}
\end{align*}
\]

The interfacial interactions are the critical factor that affects the structure of organic molecule on the surface as well as its rheological properties under confined conditions. As shown in Figure 1, the interaction between alkane and iron was derived from ab initio calculations, where the adsorption energy between C₂H₆ and iron surface was measured by Govender et al. [19].

|          | z(e) | B(eV)    | ρ(A) | C(A^6 eV) |
|----------|------|----------|------|-----------|
| O        | -0.945 | 9022.821 | 0.265 | 85.092    |
| Fe^{2+}  | 0.945 | 13032.949 | 0.190 | 0.000     |
| Fe^{3+}  | 1.4175 | 8020.285 | 0.190 | 0.000     |

Values for B, ρ and C corresponding to iron cation-oxygen and oxygen-oxygen interaction, the cation-cation interaction being described only by Columbic repulsive forces.

Figure 1. The adsorption model of C₂H₆ adsorbate on Fe(100) surface reconstructed from reference [19]. The interaction between ethane and iron surface was described using Van der Wall interaction.
The procedure to calibrate this interaction parameter was implemented using their work with L-J 9-6 potential. This parameterization was carried out at absolute zero temperature. The ethane molecule and two upper iron layers were relaxed while the rest was fixed. In order to get consistent adsorption energy with Govender’s result, the well-depth energy of iron (\(\varepsilon_{Fe}\)) for L-J 9-6 interaction between surface and lubricant was kept in range between 0.02 and 0.05 eV. The dependence of adhesive strength between ethane and iron surface on \(\varepsilon_{Fe}\) is presented in Figure 2, which indicates that the value for \(\varepsilon_{Fe}\) of 0.0425 eV is appropriate to represent the adsorption energy of \(E_{ad} = -0.23\) eV from previous ab-initio calculation [19]. The non-bond interaction parameters between iron oxide surfaces and hexadecane were conducted from the COMPASS FF and semi-ionic oxide model proposed by Zhao et al. [20] (Table 2). The molecular model was constructed with hexadecane layer allocated above Fe(100), FeO(100), and Fe\(_2\)O\(_3\)(001) surfaces. The system domain sizes \(D_x \times D_y \times D_z\) were roughly 35 \(\times\) 35 \(\times\) 60 Å\(^3\) in x, y, and z directions, respectively (Table 3). The thickness for each surface was roughly 10 Å and the initial thickness of fluid layer was 40 Å.

Table 2. Non-bond Lenard-Jones potential parameters for interaction between iron/iron oxide surfaces and hexadecane.

| Atoms  | Zhao’s work [20] | Derived from COMPASS |
|--------|------------------|----------------------|
|        | \(\varepsilon_0\) (eV) | \(r_0\) (Å) | \(\varepsilon_0\) (eV) | \(r_0\) (Å) |
| O      | 0.0034           | 3.627               | 0.0097           | 3.627 |
| Fe\(^{2+}\) | 0.01858         | 3.950               |                    | 3.950 |
| Fe\(^{3+}\) | 0.02149         | 4.077               |                    | 4.077 |
| Fe     | 0.0425\(^a\)    |                      |                    | 2.6595 |

Table 3. Domain sizes and number of hexadecane molecules for each confined shear model.

| Surfaces   | \(D_x\) (Å) | \(D_y\) (Å) | \(D_z\) (Å) | No. of C\(_{16}\)H\(_{34}\) molecules |
|------------|-------------|-------------|-------------|-------------------------------------|
| Fe         | 34.40       | 34.40       | 60.00       | 94                                  |
| FeO        | 34.98       | 34.98       | 60.00       | 94                                  |
| Fe\(_2\)O\(_3\)(001) | 34.88   | 35.25       | 60.00       | 94                                  |
| Fe\(_2\)O\(_3\)(012) | 32.50   | 40.28       | 60.00       | 94                                  |
3. Results

Surface corrugation [21-25] and adhesive strength [23, 26-28] play a vital role on molecular structure of thin hydrocarbon film. Therefore, the influences of these factors were analysed. The obtained results in Table 4 shows that both COMPASS and Buckingham FF can predict the crystalline structure of different iron compounds. Regarding the solid material, the surface structure is different from its bulk due to the propensity of atomic rearrangement after cleaving the surface. Furthermore, the unrelaxed surfaces influence significantly the interaction energy of organic molecules on solid surfaces compared to the relaxed ones [5]. Therefore, the surfaces were relaxed to reduce the residual force and optimize the surface structure. In this relaxation, the bottom layers of the surfaces were constrained to reflect its bulk structure, while the others were unconstrained. As shown in Figure 3, a 2-nm-vacuum was added on the top of surfaces. A comparison between MD, DFT and experiments in Table 5 shows the changes in inter-atomic layer spacing of iron surface with the relaxations of the top atomic layer of 1.9% and 9.4% for EAM/FS and COMPASS, respectively. This table also shows that the relaxation obtained from EAM/FS is closer with DFT calculation and experiments than COMPASS FF.

**Table 4. Lattice parameters for iron and iron oxides obtained from EAM/FS [16] for Fe, Buckingham for FeO and Fe₂O₃ [29, 30], COMPASS, and experiments**

| Compounds   | Used FF in this work | COMPASS                | Expt.            |
|-------------|----------------------|------------------------|------------------|
| Fe          | a = b = c = 2.866 Å  | a = b = 2.534 Å        | a = b = c = 2.840 Å |
|             | α = β = γ = 90°      | c = 3.583 Å            | α = β = γ = 90°  |
| FeO         | a = b = c = 4.373 Å  | a = b = c = 4.331 Å    | a = b = c = 4.33 Å |
|             | α = β = γ = 90°      | c = 13.411 Å           | α = β = γ = 90°  |
| Fe₂O₃       | a = b = c = 5.048 Å  | a = b = c = 5.055 Å    | a = b = c = 5.035 Å |
|             | α = β = γ = 90°      | c = 13.931 Å           | α = β = γ = 90°  |
|             | γ = 120°             | γ = 120°               | γ = 120°         |

*a*Lattice parameters for iron obtained from experiment in Ref. [16]

*b*Experimental lattice parameters for FeO from Ref. [29]

*c*Experimental lattice parameters for Fe₂O₃ from Ref. [30]

**Table 5. The structure of Fe(100) surface containing the top five layers after relaxation**

| Distances | Initial  | COMPASS  | EAM/FS | Ref. |
|-----------|----------|----------|--------|------|
|           | d₀ (Å)   | d (Å)    | Δd (%) | DFT | Expt. |
| Fe¹ — Fe² | 1.433    | 1.569    | 9.4    | -3.0 | -1.4 ± 3 |
| Fe² — Fe³ | 1.433    | 1.467    | 2.4    | 1.7  | 5 ± 2  |
| Fe³ — Fe⁴ | 1.433    | 1.443    | 0.7    | 1.436| 0.2   |
| Fe⁴ — Fe⁵ | 1.433    | 1.437    | 0.3    | 1.432| -0.1  |

The relaxation for FeO(100) surface using COMPASS potential is presented in Table 6 with the changes in z coordinate of the top layers are -2.2% for Fe and -8.6% for O, while there are -6.0% for Fe and 6.9% for O for Buckingham. The result indicated that the FeO(100) surface atoms were shrunk by using COMPASS FF, whereas they were expanded with Buckingham. The contraction tendency of COMPASS was consistent with that obtained by Wang et al.[33]. However, considering surface corrugation, the surface described by the Buckingham FF was smoother and more consistent with their DFT calculation. The inter-atomic space between Fe and O atoms on the top layer are df,eO = -0.9% for Buckingham potential, whereas this value is 6.4% for COMPASS. In general, there was a discrepancy of both COMPASS and Buckingham FF in comparison with DFT calculation in describing the surface structure of FeO(100), but the differences are rather small.
Figure 3. Cross-sections of Fe(100), FeO(001), Fe$_2$O$_3$(0001) slabs used in calculation. The Fe and O atoms are purple and red, respectively. Lower layers are fixed while the others are free and a vacuum of 2nm is added on the top of the surfaces.

Table 6. The structure of FeO(100) surface containing the top three layers after relaxation

| Layer 1 | Layer 2 | Layer 3 | Layer 1 | Layer 2 | Layer 3 | Layer 1 | Layer 2 | Layer 3 |
|---------|---------|---------|---------|---------|---------|---------|---------|---------|
| COMPASS | Buckingham | DFT$^a$ | COMPASS | Buckingham | DFT$^a$ | COMPASS | Buckingham | DFT$^a$ |
| $\Delta z_{Fe}/\%$ | 6.0 | 4.9 | 3.1 | -2.7 | -1.6 | $-1.2 \times 10^{-6}$ |
| $\Delta z_{O}/\%$ | 6.9 | 4.7 | 3.2 | -3.2 | -1.5 | $0.5 \times 10^{-6}$ |

For Fe$_2$O$_3$(001) surface, the inter-atomic spacing for the top surface layer moves inward 54%, whereas there is an expansion of 11.9% for the first sub-layer using Buckingham. This surface relaxation is consistent with the results obtained from DFT calculation (Table 7). In contrast, COMPASS predicts the respective contractions of 135.9% and 39.3% for the top and the first-sub surface layers, indicating that Buckingham FF is more accurate than COMPASS in describing the surface structure of Fe$_2$O$_3$. 

$^a$Result obtained from DFT calculation from Ref. [33]
Table 7. Structure of $\alpha - Fe_2O_3(0001)$ surface containing six repeat units before and after relaxation. However, only the top three repeat units are shown.

| Distances          | Initial          | COMPASS         | Buckingham | DFT          | Trainor Ref.[34] $\Delta d$(%) | Wang Ref.[35] $\Delta d$(%) | Chambers Ref.[36] $\Delta d$(%) |
|--------------------|------------------|-----------------|------------|--------------|--------------------------------|----------------------------|-------------------------------|
| $Fe^{[1]} - O^{[2]}$ | 0.846            | -0.335          | -139.5     | 0.39         | -54.0                          | -65                        | -57                           |
| $O^{[2]} - Fe^{[3]}$ | 0.846            | 0.513           | -39.3      | 0.946        | 11.9                           | 7                          | 7                             |
| $Fe^{[3]} - Fe^{[4]}$ | 0.595            | 0.556           | -6.5       | 0.261        | -56.2                          | -26                        | -33                           |
| $Fe^{[4]} - O^{[5]}$ | 0.846            | 0.928           | 9.7        | 1.071        | 26.5                           | 13                        | 15                            |
| $O^{[5]} - Fe^{[6]}$ | 0.846            | 1.099           | 29.9       | 0.942        | 11.4                           | 5                          | 5                             |
| $Fe^{[6]} - Fe^{[7]}$ | 0.595            | 0.306           | -48.5      | 0.542        | -8.9                           | -4                         | -3                            |
| $Fe^{[7]} - O^{[8]}$ | 0.846            | 0.961           | 13.5       | 0.862        | 1.9                            | 2                          | 1                             |
| $O^{[8]} - Fe^{[9]}$ | 0.846            | 0.826           | -2.4       | 0.891        | 5.3                            | 0                          | 4                             |
| $Fe^{[9]} - Fe^{[10]}$ | 0.595            | 0.681           | 14.5       | 0.567        | -4.7                           |                            |                               |
| $Fe^{[10]} - O^{[11]}$ | 0.846            | 0.749           | -11.5      | 0.868        | 2.6                            |                            |                               |

The model to investigate hexadecane adsorbed onto iron and iron oxides surfaces was carried out. The modelling lubricant consisted of a 10-Å-thick hexadecane film constituted of 40 molecules and fully covered on a square surface with the width of 35 Å for each lateral dimension (Figure 4). This adsorption simulation was carried out with a NVT simulation at room temperature 300K. During the dynamic simulations, the hexadecane molecules tended to vibrate under thermostated condition and moved toward the solid surfaces due to their non-bond interactions with the lubricant. Consequently, as presented in Figure 5, the adsorption energies increase with the simulation time and become more stable after 100 ps. The average adsorption energy of hexadecane during the last 250 ps on iron surface was smaller than those gained from iron oxides. Particularly, the respective $E_{ads}$ values of 18.5, 18.8, and 18.9 eV for Fe(100), FeO(100), and Fe$_2$O$_3$(001) surfaces. Regarding the iron oxides surfaces with interfacial interaction taken from the reference, the adsorption energy for Fe$_2$O$_3$(0001) was surface 19.3 eV – slightly larger than FeO(100) (roughly 18.8 eV). Conversely, there was a significantly larger adsorption energy on the FeO(100) surface (around 21.9 eV) compared with the Fe$_2$O$_3$(001) surface (roughly 18.9 eV) when using the COMPASS FF (Figure 5).

Figure 4. Snapshot of thin hexadecane film adsorbed on Fe(100), FeO(100), and Fe$_2$O$_3$(001) 2D-periodic slabs at 0.5 ns. The system involves 40 hexadecane molecules adsorbed on a square surface area of a 35 Å for each side dimension and 10 Å for slab thickness.
Figure 5. (i) Time evolution of adsorption energy ($E_{\text{ads}}$); and (ii) average $E_{\text{ads}}$ values of fully covered hexadecane film on Fe(100), FeO(100), and Fe$_2$O$_3$(001) surfaces. $^{ab}$ denote the model with interfacial interaction parameters obtained from COMPASS, and Buckingham FFs, respectively.

An interesting feature is that although the adsorption energies are substantially different between two implemented solid-liquid interaction FFs for FeO(100) surface, the atomic concentration profiles of hexadecane back-bone in Figure 6a are similar. In contrast, there is a moderate difference of the peak of atomic density of the first hexadecane layer on Fe$_2$O$_3$(001) surface despite the obtained adsorption energies from both FFs are close (Figure 6b). The interpretation for this observation could be due to the fact that both COMPASS and Buckingham FFs can simulate properly the surface structure of FeO(100), whereas only Buckingham FF could predict accurately the surface properties of Fe$_2$O$_3$(001). The rearrangement of surface atoms is distinct for different FFs, which result in the discrepancies in ordering structure of hexadecane lubricant on Fe$_2$O$_3$(001) surface. The separation between the first hexadecane layer and the iron surface is 3.2 Å, while it is 3.5 Å for iron oxides. This value was consistent with that obtained from Li [5] and Claire [6]. In addition, the numbers of adsorbed back-bond carbon atoms in the first layer and the saturated carbon densities in Table 8 indicate a slight difference between chosen FFs, while there is a significant difference between the surfaces with the largest number of adsorbed carbon atoms found on
FeO(100). Therefore, there is an effect of surface structure and applied FFs on the atomic density at solid-fluid interface.

**Table 8.** Computed distance between the first nearest layer of hexadecane to the surface and to the second layer, the number of saturated carbon atoms and its density, the adsorption energy as well as its mean value for each CH$_2$ group. The model with interfacial interaction parameters were obtained from: *COMPASS, aRef. [20]

| Surfaces | Distance (Å) | No. of C atoms on 1$^{st}$ layer | Saturated density (Atom/100 Å$^2$) | Adsorption energy (eV) | Adsorption energy/CH$_2$ (eV) |
|----------|-------------|-------------------------------|-----------------------------------|------------------------|-------------------------------|
| Fe       | 3.0         | 4.5                           | 184                               | 15.6                   | 15.922                        |
| FeO$^a$  | 3.5         | 4.5                           | 194                               | 16.2                   | 21.888                        |
| FeO$^b$  | 3.5         | 4.5                           | 190                               | 15.8                   | 18.792                        |
| Fe$_2$O$_3^a$ | 3.5        | 4.5                          | 192                               | 15.6                   | 18.878                        |
| Fe$_2$O$_3^b$ | 3.3       | 4.5                          | 185                               | 15.0                   | 19.337                        |

**Figure 6.** Atomic density profiles across the film thickness of hexadecane in normal direction of: (a) FeO surfaces; (b) Fe$_2$O$_3$ surfaces; (c), (d) Fe, FeO, and Fe$_2$O$_3$ surfaces. The model with interfacial interaction parameters were obtained from: *COMPASS FF, aRef. [20]
4. Conclusions

The molecular dynamic simulation had been carried out to analyse the effects of iron and its binary oxides surface to the adsorption of hexadecane on these surfaces. By employing a reliable force field for lubricant molecules, the obtained results showed a good prediction of bulk hexadecane adsorption ability. Regarding the behaviour of hexadecane on iron and iron oxide surfaces, MD results provided that it preferred to adsorb on iron oxide surface rather than its pure metal one, and layer forming was observed in these cases.

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