Monodisperse Liquid Crystalline Polymer Shells with Programmable Alignment and Shape Prepared by Seeded Dispersion Polymerization

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ABSTRACT: Monodisperse, micrometer-sized liquid crystalline (LC) shells are prepared by seeded dispersion polymerization. After polymerizing LC monomer mixtures in the presence of non-crosslinked polymer seeds, hollow LC polymer shells with programmable alignment and shape are prepared by removing the seeds. The LC alignment in the LC polymer shells can be easily manipulated by the polymer seeds, as a radial alignment is observed with amorphous poly(phenyl methacrylate) seeds and a bipolar alignment is observed with bipolar LC polymer seeds. After removal of the seeds, the radially aligned samples give radially aligned shells with small dimples. The resulting bipolar LC polymer shells collapse into a biconcave shape. Polarized optical microscopy and transmission electron microscopy indicate that the collapse occurs at the defect points in the shell. In the case of a lower crosslink density, LC polymer hollow shells with larger dimples are obtained, resulting in cup-shaped polymer particles. Biconcave LC polymer shells based on other LC mixtures have also been prepared, showing the versatility of the seeded dispersion polymerization method.

1. INTRODUCTION

Polymer shells are of interest for use in drug delivery,¹⁻³ in separation of chiral mixtures,⁴⁻⁸ and as a host for catalysts.⁹⁻¹⁰ Among hollow particles, liquid crystalline (LC) polymer shells are particularly interesting, as the alignment of LC molecules gives rise to novel anisotropic functional properties.¹¹⁻¹⁴ Monodisperse LC polymer shells have been prepared using microfluidics;¹¹⁻¹⁴ however, preparing well-defined LC polymer shells with tunable alignment with dimensions smaller than 10 μm remains a challenge.

Herein, we report the facile preparation of monodisperse micrometer-sized LC polymer shells with programmed alignment and shape via seeded dispersion polymerization. Since the outer surface of the seed functions as an alignment layer for the LC molecules making up the shell, we expected differing alignments in the shell dependent on whether amorphous poly(phenyl methacrylate) (PPhMA) or LC bipolar polymer particles were used as seeds. In what follows, we show that the nature of the seed indeed determines the alignment and shape of the monodisperse, micrometer-sized shell, which consists of a combination of the acrylic LC monomers as shown in Figure 1.

2. EXPERIMENTAL SECTION

2.1. Materials. 4-Methoxyphenyl 4-((6-(acryloyloxy)hexyl)oxy)benzoate (1), 2-methyl-1,4-phenylene bis[4-((6-(acryloyloxy)hexyl)oxy)benzoate] (2), 4,4’-bis(6-acryloyloxyhexyloxy) azobenzene (3), and 4-(6-acryloxyhexyl-1-oxy)benzoic acid (4) (see Figure 1) were purchased from Synthon Chemicals, Germany. Phenyl methacrylate was purified according to a well-established procedure.¹²¹³ Phenyl methacrylate was purified according to a well-established procedure.¹²¹³ 4-(6-Acryloxyhexyl-1-Oxy)benzoic acid (4) was prepared by a reaction of 4-Methoxyphenyl 4-((6-(acryloyloxy)hexyl)oxy)benzoate (1) with 2,4-dimethylpentane and 4-(acryloyoxy)hexyloxybenzoate (2) and 2,4-dimethylpentane. The resulting product was purified by column chromatography on silica gel using a mixture of methanol and dichloromethane as eluent. The purity of the product was confirmed by ¹H and ¹³C NMR spectroscopy. Other reagents and solvents were used as received.

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PhMA, 97%) was purchased from TCI Europe. Polyvinyl pyrrolidone (PVP K30) and 2,2′-azobis(2-methylpropionitrile) (AIBN, 98%) were purchased from Sigma-Aldrich. All solvents were purchased from Biosolve.

2.2. Preparation of LC Shells via Seeded Dispersion Polymerization. 2.2.1. Synthesis of Core−Shell LC Particles with Amorphous PPhMA Seeds. Poly(phenyl methacrylate) (PPhMA) seeds were prepared by dispersion polymerization. To this end, 100 mg of PVP and 10 mg of AIBN were added to a 50 mL round-bottom flask, which was then evacuated and backfilled with nitrogen 3 times. Subsequently, 1 mL of PhMA dissolved in 9 mL of ethanol was added to the flask, after which the flask was put into an oil bath preheated to 70 °C and stirred at 200 rpm under N2 overnight. After the polymerization, the PPhMA particles were centrifuged and washed with ethanol three times. Yield = 77%.

The PPhMA seed particles were then used to prepare core−shell LC particles. To a 50 mL round-bottom flask, 90 mg of monomer 1, 5 mg of monomer 2, 5 mg of monomer 3, 20 mg of PVP, and 2 mg of AIBN were added, after which the round-bottom flask was then pumped and backfilled with nitrogen 3 times. Subsequently, 1 mL of PhMA dissolved in 9 mL of ethanol was added to the flask, after which the flask was put into an oil bath preheated to 70 °C and stirred at 200 rpm under N2 overnight. After the polymerization, the PPhMA particles were centrifuged and washed with ethanol three times. Yield = 77%.

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Figure 1. (a) Monomers employed to prepare the LC polymer shells. (b) Schematic representation of the synthesis of LC shells with radial alignment using amorphous PPhMA seeds. (c) Schematic representation of the synthesis of LC shells with bipolar alignment using bipolar LC seeds.
was carried out for another 4 h. After polymerization, the dispersion was centrifuged and washed with ethanol to obtain the particles. The core–shell particles were stored in 5 mL of ethanol prior to use, and the solid content was accurately measured by drying 50 μL of suspension in a DSC pan. Yield = 61%, where yield is defined as the weight of the core–shell particles after seeded dispersion polymerization divided by the total weight of LC monomers.

2.2.4. Removal of the Seeds. The particle suspension (1 mL) of predetermined solid content was added to 9 mL of tetrahydrofuran (THF), and the particles were allowed to sediment. The solvent was then poured out, and fresh THF was added. This process was repeated 3 times in total. Then, 10 mL of ethanol was added dropwise with agitation and the particles were allowed to sediment. The solvent was then poured out, and 10 mL of fresh ethanol was added dropwise with agitation. This process was repeated 3 times in total. The LC polymer shells were stored in 5 mL of ethanol prior to use, and the solid content was accurately measured by drying 50 μL of suspension in a DSC pan. Weight loss = 52% (entry 1), 63% (entry 3), and 36% (entry 5), where weight loss = 1 − (weight of THF-treated particles/ weight of initial particles); the weights were calculated from the solid content and the volume of the corresponding suspensions.

2.2.5. Photoresponsiveness of the Bipolar LC Polymer Shell. LC polymer shells in chlorobenzene (0.3 mg/mL) were added in a quartz cuvette. The suspension was immediately subjected to UV–vis measurement, denoted as “measurement 0.” Then, the suspension was irradiated with 365 nm light for 1 min, and the suspension was subjected to another UV–vis measurement, denoted as “measurement 1.” The suspension was irradiated with 455 nm light for 1 min, and the suspension was subjected to another UV–vis measurement, denoted as “measurement 2.” Five cycles of alternative irradiation were performed, and the UV–vis spectra were measured.

2.3. Characterization. Polarized optical microscopy (POM) images were taken with a Leica CTR6000 polarized optical microscope and a Leica DFC 420C camera. POM images were taken by drying the particle suspensions on clean glass slides unless otherwise stated. 1H NMR spectra were recorded on a 400 MHz Bruker Avance III HD spectrometer in deuterated chloroform with tetramethyl silane used as an internal standard. Scanning electron microscopy (SEM) images were taken with a JEOI TM 220 A. The average size and coefficient of variation of the particle size distribution (Cv = σ / μ) were measured using ImageJ. The polymer particles and their cross sections were investigated via transmission electron microscopy (TEM) using a Tecnai 20 (type Sphera) by FEI operating with a LaB 6 filament at 200 kV under slight under-focus conditions. For the cross section, particles were embedded in an EPOFIX epoxy medium. Cross sections were cut at room temperature using an ultramicrotome (Reichert-Jung Ultracut E) with a setting thickness of 60 nm. The obtained cross sections were transferred to a carbon film-covered grid (Electron Microscopy Sciences, CF200-CU). UV–vis spectra were measured with a Shimadzu UV-3102 PC. Thorlabs light emitting diode lamps with collimation adapters were used to provide 365 and 455 nm lights. The distance between the lamps and the cuvette was about 30 cm. The light intensities were about 120 mW/cm² (365 nm) and 45 mW/cm² (455 nm).

3. RESULTS AND DISCUSSION

3.1. Synthesis of LC Shells via Seeded Dispersion Polymerization with PPhMA Seeds. Monodisperse PPhMA particle seeds were first prepared by dispersion polymerization at 70 °C in ethanol, with PVP as the stabilizer and AIBN as the thermal initiator. SEM images revealed an average diameter of 1.50 μm and a coefficient of variation of 3.7% (Figure 2a). The PPhMA particle seeds were redispersed in ethanol and then added to an ethanol solution containing LC monoaacrylate 1, LC diacrylate 2, and azobenzene diacrylate 3 (Table 1, entry 1, Figure 2b), PVP, and AIBN. The suspension was heated to 70 °C and polymerized overnight, yielding spherical core–shell particles with an average diameter of 1.69 μm and a coefficient of variation of 3.8% (Figure 2b). This indicates a thickness of approximately 100 nm for the cross-linked shell. To remove the PPhMA core (which consists of linear PPhMA chains), the particles were treated with THF repeatedly, prior to transferring and storing in ethanol. The treatment with THF resulted in a weight reduction of 52%, which suggests a near complete removal of the PPhMA core; the original composition of the core–shell particles was 56 wt % core and 44 wt % shell. SEM images of the LC shells (Figure 2c) show dimpled particles with an average diameter of about 1.46 μm. The Hansen solubility parameter of the LC polymer shells was estimated with a group contribution method, and the solubility distance (Rs) to ethanol and THF was calculated (Tables S1 to S3). The lower number of Rs to THF suggests that THF has a higher compatibility than ethanol with the LC, and hence, it is conceivable that THF diffusion out of the shell is faster than ethanol diffusion into the shells, resulting in an osmotic pressure that could locally compress the shell. The TEM images of a cross section of the shells (Figure 2d) confirm a thickness of about 100 nm for the LC shells, which can be seen as the dark rings with dimples; this result is consistent with the increase of the particle diameter after the seeded dispersion polymerization (Figure 2a,b).

To study the LC alignment, the LC shells were dispersed in ethanol and dried on a glass slide for POM. The POM image in Figure 3a without crossed polarizers shows monodisperse, micrometer-sized, almost spherical shells with occasional dimples consistent with the SEM image in Figure 2c. The POM images obtained with crossed polarizers are shown in Figure 3b,c. Maltese crosses were always observed whose direction was always parallel to the orientation of one of the crossed polarizers, indicating that the LC molecules in the polymer shells aligned radially, that is, perpendicular to the shell surface, as schematically shown in Figure 3d.
A small volume of suspension was withdrawn from the 2 h to form the noncrosslinked bipolar LC seeds in situ (Figure 3). The orientation of the crossed polarizers is indicated by the Maltese crosses observed in POM images, again indicating bipolar alignment of these LC particles and the conversion, respectively. The LC particle seeds showed bright spots under crossed polarizers, indicating that the LC molecules in the particles aligned in a bipolar manner, similar to that reported before. The integration of remaining acrylate peaks at 5.5 to 6.5 ppm in the 1H NMR spectrum indicated that around 30% of monomer 1 remained unreacted at this point (Figure S1). Therefore, the amount of monomer 1 added in the second stage was correspondingly reduced to maintain the composition of the shells identical to the shells made using the PPhMA seeds; the LC monomer mixture was added in the flask without separation and purification of the LC seeds. Seeded dispersion polymerization was continued overnight, yielding core–shell particles with an average diameter of 1.63 μm and a coefficient of variation of 4.3% (Table 1, entry 3, and Figure Sb). The noncrosslinked LC seeds were removed by THF, similar to the removal of the PPhMA seeds. The weight of the LC shells was reduced by 63%, suggesting successful seed removal, as the core and shell initially made up 57 and 43% of the core–shell particle. SEM images revealed that the remaining hollow shells collapsed into a biconcave shape with two dimples distributed symmetrically at opposite faces of the hollow shells (Figure Sc). While the average diameter of the shells remained almost unchanged at 1.59 μm, the height of the shell decreased significantly to 0.61 μm. TEM images further confirmed the size and shape of the LC shells, and the symmetrically distributed dimples were observed as a relatively brighter center upon tilting the sample (Figure 5d to f).

Maltese crosses were observed parallel to the directions of the cross polarizers after drying the shells on a glass slide (Figure 6a to c). However, since the biconcave shells uniformly “lie down” on the glass slide, meaning that the radii of the spheroids are coplanar with the glass slide surface, in order to investigate the birefringence from different angles, the shells were suspended in glycerol and filled in a glass cell for POM (Figure 6d to f). Due to the flow of the solvent, LC shells first rotated and flowed (see Movie S1), and as a result, the shape and birefringence pattern of the LC shells continuously changed. Static POM images were taken when the shells stopped moving, and some shells showed dark lines parallel to the direction of the collapse rather than the Maltese cross. From the POM images, it was concluded that the LC shells had a bipolar alignment, and the collapse occurred at the defects, which is probably because the unaligned defects have a lower modulus than the aligned regions (Figure 6g).

LC shells prepared with bipolar LC seeds and a LC monomer mixture with a lower crosslink density (Table 1, entry 4, Figure 7) showed cup-shaped particles with large dimples, similar to the shells prepared with PPhMA seeds using the same LC mixture (Table 1, entry 2, Figure 4). Moreover, Maltese crosses were observed in POM images,

Table 1. LC Polymer Core–Shell Particles and Hollow Shells Prepared by Seeded Dispersion Polymerization

| entry | monomer mixture | type/diameter of the seeds | diameter of the core–shell particles | diameter/shape of the LC shells | alignment of the LC shells |
|-------|----------------|---------------------------|-------------------------------------|-------------------------------|---------------------------|
| 1     | 1, 2, 3        | PPhMA/1.50 μm             | 1.69 μm                             | 1.46 μm/spheres with small dimples | radial                     |
| 2     | 1, 3           | PPhMA/1.50 μm             | 1.86 μm                             | N.A. /cup-shape              | radial                     |
| 3     | 1, 2, 3        | bipolar LC polymer seeds* | 1.83 μm                             | 1.59 μm × 0.61 μm/biconcave   | bipolar                    |
| 4     | 1, 3           | bipolar LC polymer seeds* | 1.82 μm                             | N.A. /cup-shape              | radial                     |
| 5     | 1, 2, 4        | bipolar LC polymer seeds* | 1.38 μm                             | 1.32 μm × 0.68 μm/biconcave   | bipolar                    |

*No diameters were measured because the seeds were not isolated. The diameter of the LC shells cannot be measured accurately due to the extensive deformations.

Seeded dispersion polymerization of a LC monomer mixture consisting of monomers 1 and 3 was also performed (Table 1, entry 2, and Figure 4). Since there are relatively more monoacrylates in the forming shells than in entry 1, the final crosslink density in the polymerized shells is likely lower, resulting in hollow shells capable of collapsing and forming larger dimples and cup-shaped polymer particles. Maltese crosses were observed in the POM images, again indicating radial alignment of the LCs in the polymer shells.

**3.2. Synthesis of LC Shells via Seeded Dispersion Polymerization with Bipolar LC Seeds.** To induce noncentral symmetric alignment in the LC shells, dispersion polymerization of LC monomer 1 developed by Zentel et al. was adopted to produce noncrosslinked bipolar LC polymer seeds, which were then used as seeds. Upon forming the bipolar LC polymer seeds, the monomer mixture for the LC polymer shell was added to the dispersion without separation of the seeds, and the polymerization was continued to form the core–shell particles. LC monomer 1, PVP, and AIBN were dissolved in ethanol and polymerized at 70 °C for 2 h to form the noncrosslinked bipolar LC seeds in situ (Figure 5a). A small volume of suspension was withdrawn from the mixture and subjected to POM and 1H NMR to determine the alignment of these LC particles and the conversion, respectively. The LC particle seeds showed bright spots under crossed polarizers, indicating that the LC molecules in the particles aligned in a bipolar manner, similar to that reported before. The integration of remaining acrylate peaks at 5.5 to 6.5 ppm in the 1H NMR spectrum indicated that around 30% of monomer 1 remained unreacted at this point (Figure S1). Therefore, the amount of monomer 1 added in the second stage was correspondingly reduced to maintain the composition of the shells identical to the shells made using the PPhMA seeds; the LC monomer mixture was added in the flask without separation and purification of the LC seeds. Seeded dispersion polymerization was continued overnight, yielding core–shell particles with an average diameter of 1.63 μm and a coefficient of variation of 4.3% (Table 1, entry 3, and Figure Sb). The noncrosslinked LC seeds were removed by THF, similar to the removal of the PPhMA seeds. The weight of the LC shells was reduced by 63%, suggesting successful seed removal, as the core and shell initially made up 57 and 43% of the core–shell particle. SEM images revealed that the remaining hollow shells collapsed into a biconcave shape with two dimples distributed symmetrically at opposite faces of the hollow shells (Figure Sc). While the average diameter of the shells remained almost unchanged at 1.59 μm, the height of the shell decreased significantly to 0.61 μm. TEM images further confirmed the size and shape of the LC shells, and the symmetrically distributed dimples were observed as a relatively brighter center upon tilting the sample (Figure 5d to f).

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indicating that the LC molecules aligned radially in the shells, even with bipolar LC seeds. This result suggests that an appropriate crosslink density is necessary to obtain the bipolar alignment in the shells.

### 3.3. Seeded Dispersion Polymerization of Other LC Mixtures

Seeded dispersion polymerization was also performed using a nematic LC mixture containing monomers 1, 2, and 4 (weight ratio = 65/15/20, Table 1, entry 5). Monomer 4 (Figure 1a) is chosen to investigate the versatility of this seeded dispersion polymerization approach, since carboxylic acid-functionalized monomers have been known to significantly undermine the colloidal stability and make dispersion polymerization challenging.\(^22\) The in situ preparation of the bipolar LC seeds of monomer 1 was performed as previously described, but since carboxylic acid groups significantly undermine the colloidal stability and are known to make dispersion polymerization challenging,\(^22\) the second stage of the seeded dispersion polymerization was modified, with the amount of PVP increased and the polymerization time reduced to 4 h. SEM images showed that the average diameter of the resulting core–shell particles was 1.38 μm and the coefficient of variation was 8.1% (Figure 8a), indicating that
Figure 6. POM images of the LC shells (a) without and (b and c) with crossed polarizers (scale bar = 10 μm) prepared by using bipolar LC seeds. POM images of the LC shells suspended in glycerol (d) without and (e and f) with crossed polarizers (scale bar = 10 μm). The orientation of the crossed polarizers is indicated by the arrows. (g) Schematic representation of the structure and LC alignment of the LC shells.

Figure 7. SEM images of (a) core−shell particles and (b) LC shells (scale bar = 2 μm). POM images of LC shells (c) without and (d and e) with cross polarizers (scale bar = 10 μm). The orientation of the cross polarizers is indicated by the arrows.
stable dispersion and particles with low polydispersity were obtained, even in the presence of the carboxylic acid-functionalized LC monomers. The core-shell particles were smaller than the previous core-shell particles (Table 1, entry 3), probably because the yield is lower as the polymerization time is shortened. After THF treatment, a weight loss of 36% was observed, while the core and shell initially made up 68 and 32% of the core-shell particle. This indicates that the removal of the seeds is not complete, probably because the diffusion of the polymer chains out of the particles was slowed by the dense LC shell, and harsher conditions are needed to remove the LC seeds more completely. Similar birefringence patterns to those seen in the previous bipolar shells were observed with POM (Figure 8c to h), confirming bipolar alignment in these LC shells, and that effective control on the LC alignment is feasible with the seeded dispersion polymerization method even with more challenging LC monomers.

4. CONCLUSIONS

We report the successful preparation of monodisperse, micrometer-sized LC polymer shells via seeded dispersion polymerization. Investigation of the chemical structure of the seeds revealed that with non-LC (PPhMA) seeds, the molecules in the LC polymer shells aligned radially, and shells with small dimples are formed. By polymerizing a monomer mixture with an appropriate crosslinking density in the presence of bipolar LC seeds, hollow bipolar LC polymer shells can be prepared, which collapsed in a biconcave shape. For lower crosslink density LC polymers, cup-shaped polymer shells are formed. LC polymer shells with other functional groups can also be prepared with this method, revealing the versatility of the seeded dispersion polymerization method.

Our findings disclose a new method to fabricate monodisperse LC polymer shells with programmable alignments and shapes. Preliminary results reveal that the azobenzene moieties in the shells can be photoswitched by alternative irradiation of 365 and 455 nm light (Figure S2),
which might be interesting for drug delivery and soft actuator applications.

**ASSOCIATED CONTENT**

Supporting Information
The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.macromol.1c00884.

Hansen solubility parameter of the liquid monomers and the solubility distance to ethanol and THF; NMR spectrum of the linear LC seeds using monomer 1; and UV−vis data of the LC polymer shells in entry 3 suspended in chlorobenzene after 365 and 455 nm light irradiation (PDF).

Rotation and flow of LC shells due to the flow of the solvent (Movie S1) (MP4).

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The authors declare no competing financial interest.

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