Effect of Yb concentration on the structural, magnetic and optoelectronic properties of Yb doped ZnO: first principles calculation

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Abstract
Density functional theory-based investigation of the electronic, magnetic, and optical characteristics in pure and ytterbium (Yb) doped ZnO has been carried out by the plane-wave pseudopotential technique with generalized gradient approximation. The calculated lattice parameters and band gap of pure ZnO are in good agreement with the experimental results. The energy band gap decreases with increasing Yb concentration. The Fermi level moves upward into the conduction band after doping with Yb, which shows the properties of an n-type semiconductor. New defects were created in the band-gap near the conduction band attributed to the Yb-4f states. The magnetic properties of ZnO were found to be affected by Yb doping; ferromagnetic property was observed for 4.17% Yb due to spin polarization of Yb-4f electrons. The calculated optical properties imply that Yb doped causes a blue shift of the absorption peaks, significantly enhances the absorption of the visible light, and the blue shift of the reflectivity spectrum was observed. Besides, a better transmittance of approximately 88% was observed for 4.17% Yb doped ZnO system. The refractive index and the extinction coefficient were observed to decrease as the Yb dopant concentration increased. As a result, we believe that our findings will be useful in understanding the doping impact in ZnO and will motivate further theoretical research.

Keywords First principles · DFT + U · Doped ZnO · Electronic structure · Magnetic properties · Optical properties

1 Introduction
Recently, studies are increasingly focusing on developing the ZnO semiconductor as a functional material for optoelectronic and electronic devices. As a wide band gap semiconductor about 3.37 eV and a large exciton binding energy (∼60 meV) at room temperature, ZnO can be used in many potential applications such as blue and ultraviolet (UV) lasers, light-emitting diodes, Spintronics and transparent electrodes in solar cells (Look
et al. 1998; Look 2006; Namgung et al. 2019). Because doping ZnO is one of the most significant issues in order to design ZnO based devices, rare-earth (RE) doped ZnO has become recently the focus of numerous studies owing to their unique optical properties and promising applications in optoelectronic devices (Röder et al. 2019; Chen et al. 2019; López-Mena et al. 2021). Rare earth generates great interest due to the 4f-4f intrashell transitions, which provide intense emission lines in the UV, visible, and infrared (IR) regions and improve luminescence efficiency through energy transfer processes (Daksh and Agrawal 2016). Because it is one of the rare earth elements, Yb can improve the optical and electrical properties of ZnO. Many experimental studies regarding the influence of Yb doping content on the structural, optical and electrical properties of ZnO have been reported. Cerrato et al. (Cerrato et al. 2020) investigated the role of the Yb dopant in the photoactivity of ZnO, finding that with a small amount of Yb, the photoactivity of ZnO was enhanced. Heng et al. (Heng et al. 2020) found that for concentrations down to 0.1 at.%. of Yb, the near band-edge emission PL of pure ZnO was increased. In another study (Heng et al. 2017), he found that the ultra violet emission of the sputtered ZnO increased 100 times after introducing 5 at.% of Yb. A recent systematic study conducted by López-Mena et al. (López-Mena et al. 2020) investigated the effect of ytterbium doping on the structural, morphological and optical properties of ZnO thin films, their findings suggest that Yb-doped ZnO thin films are suitable for optoelectronic devices, solar cells. Given the large number of experimental studies on Yb-doped ZnO, there is still very limited theoretical research on the structural and optical properties of Yb-doped ZnO. For example, Khuili et al. (Khuili et al. 2020) reported a comparative study on the structural, optical and electrical properties of rare earth (Yb, Tm, Ce) doped ZnO, finding that the band gap increased after doping and the Fermi level has been shifted to the conduction band, revealing n-type characters. In this regard, this study aims to develop a theoretical framework for understanding the effect of Yb concentration on the electronic structure, magnetic and optical properties of ZnO using the first-principles density functional theory. We believe that this work could support the future design of ZnO materials by providing significant insight into the microscopic characteristics of the materials, and will also encourage further theoretical investigations.

2 Models and calculation method

Pure ZnO exhibits a hexagonal wurtzite structure that belongs to the P63mc space group and has a symmetry of C6v-4. Analysis was performed using Yb concentrations of 4.17, 6.25, and 12.5 at.%. and the structure of the doped ZnO systems was hexagonal wurtzite. Based on the primitive cell, a (3×2×2), (2×2×2) and (2×2×1) ZnO supercells were constructed as ZnO models. Figure 1a shows a pure ZnO, Fig. 1b shows Yb-doped ZnO, which corresponds to Zn$_{0.958}$Yb$_{0.0417}$O. Figure 1c, on the other hand, shows Yb doped ZnO which corresponds to Zn$_{0.9375}$Yb$_{0.0625}$O. Figure 1d, on the other hand shows a Yb doped ZnO which corresponds to Zn$_{0.875}$Yb$_{0.125}$O where one Yb atom substituting one of the Zn atoms. These models correspond to 4.17 at.% 6.25 at.% and 12.5 at.% of Yb in ZnO.

The Cambridge Serial Total Energy Package (CASTEP) module, which is based on density functional theory (DFT), was used to do all computations in this work (Clark et al. 2005). The generalized gradient approximation with the generalized gradient approximation of Perdew-Burke-Ernzerhof (Perdew et al. 1996) (GGA-PBE) and the GGA+U approximation were used to perform geometry optimization and energy calculations for all systems. Structural...
optimization was performed on each model prior to calculating their electronic and optical properties. Moreover, the plane-wave ultrasoft pseudopotential method (Vanderbilt 1990) with valence electron configurations of 4s^23d^{10} for Zn, 2s^22p^4 for O, and 4f^{12}5s^25p^66s^2 for Er was used to describe the interactions between the ionic core and valence electrons and allow the present calculation to be performed with a cut-off energy of 500 eV (Ma et al. 2013). The Brillouin zone integrations generated, automatically by the Monkhorst–Pack method were performed over a 5×5×4 grid size for pure ZnO (Harun et al. 2017) and 4×4×2 grid size for Yb-doped ZnO (Li et al. 2011). The DFT + U method was adopted to describe the electronic structures accurately, by applying Hubbard U correction to the Zn-3d states and O-2p states. According to Wu et al. (2013), the U_{p,O} value of 7 eV for oxide materials is appropriate for first principles calculations. Zn-3d and O-2p’s U values were set to 10.00 and 7.00 eV, respectively, while Yb-4f’s U value was tuned to 6.00 eV. The self-consistent iteration convergence threshold was set at 10^{-5} eV/atom, and all lattice constants and atomic locations for each supercell were totally relaxed until the maximal force on each atom was less than 0.03 eV/Å, the internal stress was below 0.05 GPa, and the displacement of each atom was below 0.01 Å.

The optical properties of the studied systems including absorption coefficient \( \alpha(\omega) \) and reflectivity \( R(\omega) \) can be obtained on the basis of complex dielectric function \( \varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) \) where \( \varepsilon_1(\omega) \) and \( \varepsilon_2(\omega) \) are the real and imaginary part which can be calculated from the Kramers–Kronig’s equation (Li et al. 2009) as follows:

\[
\varepsilon_1(\omega) = 1 + \frac{2\varepsilon_0}{\pi} \rho_0 \int \frac{\omega \varepsilon_2(\omega) d\omega}{\omega^2 - \omega^2}
\]

\[
\varepsilon_2(\omega) = \frac{2\varepsilon_0^2}{\Omega \varepsilon_0} \sum_{k,v,c} |\psi_k^c|^2 |\tilde{u}r|^2 |\psi_k^v|^2 \delta(E_k^c - E_k^v - E)
\]

Fig. 1  Schematic of the structure of ZnO: (a) ZnO; (b) 4.17% Yb-; (c) 6.25% Yb; (d) 12.5% Yb
where $\omega$ is the incident photon frequency, $\rho_o$ is the polarization response, $\hat{u}$ is the vector defining the polarization of incident electric field, $e$ is the electronic charge, and $\Omega$ is the unit cell volume. The subscripts C and V refer to the conduction and valence bands, respectively.

The transmittance was calculated from the absorption coefficient and reflectivity and can be given by:

$$T = (1 - R)^2 e^{-\alpha d}$$

where $T$ represents transmittance, $\alpha$ and $R$ represent respectively the absorption coefficient and reflectivity, and $d$ represents the film thickness which was set at 250 nm in this study.

## 3 Results and discussion

### 3.1 Structural properties

In order to evaluate the validity of the experimental data, we firstly performed an optimization of the optimum structures of pure ZnO, Yb doped ZnO with various Yb concentrations. The calculated lattice parameters $a$, $c$ and $c/a$ for the optimized pure ZnO are respectively, $a = 3.249 \, \text{Å}$, $c = 5.216 \, \text{Å}$, and $c/a = 1.605$ which are in good agreement with the experimental values [(Farhat et al. 2018; Bouhouche et al. 2018; Naik et al. 2020; Achehboune et al. 2021), and in good agreement with other theoretical calculations (Khuili et al. 2020; Meng et al. 2017), indicating that our calculations are reliable and prove the validity of our model.

Table 1 shows a summary of the structural parameters of ZnO doped with different concentrations of Yb. It was observed that the lattice parameters $a$ and $c$ increased with increasing Yb concentration, and consequently, the unit cell volume was increased. This result is consistent with another theoretical result (Khuili et al. 2020), as well as the successful preparation in experimental results (Senol 2016). This can be due to the difference in ionic radius between Yb (0.87 Å) and Zn (0.74 Å), contributing to crystal lattice expanding due to the distortion of the crystal lattice, the increase in volume cell is due to the repulsion between Zn and Yb, which produces additional cell volume expansion. The total energy values showed that the geometric configuration of the pure ZnO is that of the most stable structure. The Zn–O bond length decreased slightly with increasing Yb content, whereas the Yb-O bond length decreased with increasing Yb concentration. Furthermore, the Mulliken charge determines the amount of electron density sheared within such a crystal lattice by an atom, in which more positive values mean that more electrons are contributed by the associated atom. This indicates that the Yb dopant contributes more electrons than Zn because of the Mulliken charge values for Yb that are more positive than those of Zn, and the actual values of the charge interaction Yb-O are higher than those of Zn–O.
3.2 Electronic properties

3.2.1 Band structure

To study the effect of Yb doping concentrations, we have firstly calculated the band structure of pure ZnO. Figure 2a shows the calculated band energy structure of pure ZnO. As observed, pure ZnO exhibits a direct band gap; the valence band maximum and the conduction band minimum are located at the G point of the Brillouin zone. An estimated 0.74 eV band gap of pure ZnO was achieved using only DFT calculations, which is far from the experimental value of 3.37 eV (Özgür et al. 2005). In several studies (Yao et al. 2012; Xia et al. 2014; Farooq et al. 2016; Wen et al. 2018), the estimated energy band gap of ZnO using only GGA is between 0.74 and 0.81 eV. The underestimation of the band gap is provided by the strong p-d coupling. To correct this underestimation, we used GGA + U to affect the inaccurate description of strongly localized electrons. Therefore, the GGA + U method can help us to obtain more accurate optical properties. The calculated optical band gap for pure ZnO is 3.38 eV, which is in line with results reported in the experiment and other theoretical calculations [Ma et al. 2013; Obeid et al. 2019; Wang et al. 2020). The results of the band structures of pure and Yb doped ZnO with various concentrations are illustrated in Fig. 2. It can be observed from the Fig. 2b–d that the structures have a direct band gap whose value decreases with increasing Yb amount. It has been determined that the band gaps of $\text{Zn}_{0.958}\text{Yb}_{0.0417}\text{O}$, $\text{Zn}_{0.937}\text{Yb}_{0.0625}\text{O}$ and $\text{Zn}_{0.875}\text{Yb}_{0.125}\text{O}$, are 3.24, 3.19 and 3.22 eV, respectively, which are smaller than pure ZnO. This is in contrast to earlier experimental findings of Ref. (López-Mena et al. 2020), these results attributed the increase in the optical band to the reduction in the Zn–O bond lengths. Moreover, new occupied states appeared between the conduction band and the Fermi level attributed to the 4f orbitals of the Yb dopant, which are slightly shifted to lower energy as we increase the Yb concentration. It is also clear that the Yb-doped ZnO system is an n-type degenerate semiconductor because of the Fermi level that enters the conduction band to varying degrees and also due to the donor states created close to the conduction bands. A similar behavior was observed in (Daksh and Agrawal 2016; Wen et al. 2018; Wang et al. 2020). The introduction to Yb ions in ZnO system caused the band gap to increase due to the Burstein–Mott effect (Burstein 1954; Moss 1954).

### Table 1 Calculated cell parameters, volume, total energy, bond length, and Mulliken charges of pure ZnO, and ZnO doped with various Yb concentrations

| Cell parameters | Volume (Å³) | Total energy (eV) | Bond length (Å) | Mulliken charges |
|-----------------|-------------|-------------------|----------------|-----------------|
| a (Å)           | c (Å)       |                   | Zn–O           | Yb–O            | Zn    | Yb   | O    |
| Pure ZnO        | 3.249       | 5.216             | 47.719         | −4292.7         | 2.00517| −0.81| −0.81|
| $\text{Zn}_{0.958}\text{Yb}_{0.0417}\text{O}$ (4.17%)| 3.287       | 5.262             | 49.235         | −4685.2         | 2.02700| 0.82| 0.80| −0.81|
| $\text{Zn}_{0.937}\text{Yb}_{0.0625}\text{O}$ (6.25%)| 3.309       | 5.351             | 50.761         | −4880.54        | 2.04431| 0.81| 0.79| −0.81|
| $\text{Zn}_{0.875}\text{Yb}_{0.125}\text{O}$ (12.5%)| 3.343       | 5.378             | 52.08          | −5466.38        | 2.03831| 0.82| 0.78| −0.80|
3.2.2 Density of states

Further insight into the contributions of different states to the energy bands is given by the total state density (TDOS) and partial state density (PDOS). Figure 3 presents the total density of states (TDOS) and the partial density of states (PDOS) of the pure ZnO and Yb-doped ZnO system. It can be observed from Fig. 3a that the valence band of pure ZnO is mainly composed of the 3d state of Zn and the 2p state of O and the conduction band is mainly composed of the 4s state of Zn and the 2p state of O. The valence band located between −15.5 and 0 eV is divided into three parts. Within the range of −5.4 to 0 eV, it is mainly composed of O 2p with a minor contribution from the Zn 3d states, resulting in d-p coupling. On the other hand, the valence bands within the range of 8.7–5.4 eV are mainly composed of Zn 3d and partial O 2p states. Lastly, an isolated band in the range of −15.5 to −13.8 eV at the valence band is formed with O 2s state. The composition of conduction bands, on the other hand, is consistent with the total density of ZnO in that they are mainly formed by the Zn 4s and O 2p states. For ZnO doped with Yb shown in Fig. 3b–d, we observe that the valence band shifted towards low energies compared to pure ZnO and continues to shift with increasing amount of Yb. It was also observed that the PDOS of Yb-doped ZnO systems present occupied states at near Fermi level, which come mainly from the Yb-4f orbital. Moreover, the conduction band mainly results from Yb-5d and O-2p orbitals. Consequently, the band structure and DOS show that the increase of Yb doping

Fig. 2. Band structures of pure ZnO and ZnO doped systems, where the Fermi level is set to 0: a ZnO; b 4.17% Yb; c 6.25% Yb; d 12.5% Yb
concentration causes the Fermi level to shift to the conduction band and the DOS to move to lower energy, and the existence of occupied states around the Fermi level will basically determine the optoelectronic characteristics of these materials, indicating that doping with Yb can obtain a high-quality n-type ZnO.

### 3.3 Magnetic properties

The partial density of states (PDOS) with spin polarization was performed for all materials in order to illustrate the magnetic mechanism and the effect of Yb as dopant in ZnO with different Yb concentrations. As shown in Fig. 4, the spin-up and spin-down of the total density of states of undoped ZnO were symmetric, indicating that ZnO is a non-magnetic material; the calculated total magnetic moment of pure ZnO is 0.00018 \( \mu_B \). However, for 4.17% of Yb-doped ZnO it is noticed that the spin-up and spin-down channels were asymmetrical and show that Yb-doped ZnO has a ferromagnetic property with an optimal concentration of 4.17%, which exhibiting magnetism behavior in ZnO-Yb\(_{25}\) system. It is clear from the figure that there is a clear spin polarization between the partial DOS of the two spin channels near the Fermi level due to spin polarization of 4f electrons of Yb atom (inset image); the total magnetic moment was found to be increased to 0.048 \( \mu_B \); the ferromagnetism found in Yb doped ZnO is originated from the ferromagnetic coupling between Yb atoms controlled by the closest oxygen atoms. However, as the concentration increased.

Fig. 3  TDOS and PDOS of pure ZnO and ZnO doping systems, where the Fermi level is set to 0: (a) ZnO; (b) 4.17% Yb-; (c) 6.25% Yb; (d) 12.5% Yb
up to 6.25%, the spin up and spin down became symmetrical, and the magnetic moment of the doped ZnO structure was decreased; this behavior can be explained by the increase in the antiferromagnetic superexchange interaction along with Yb doping as a result of the decrease in the distance between the Yb ions. In addition, these two system show an n-type degenerate semiconductor as shown from Fig. 2. The calculated total magnetic moment became 0.0087, 0.0052 μB for 6.25% and 12.5% respectively. These current findings show the characteristics of diluted magnetic semiconductors (DMSs) and can help in the design and preparation of new ZnO-based magneto-optoelectronic applications.

### 3.4 Optical properties

#### 3.4.1 Dielectric function

The imaginary part of the dielectric function is a critical characteristic of the optical properties of any material, which can explain the energy required for the electrons to transit from the valence band to the conduction band. Figure 5 illustrates the imaginary parts of the dielectric functions of pure ZnO, Yb-doped ZnO with different concentrations of Yb. As can be seen in Fig. 5, pure ZnO has three main peaks in ε₂(ω), which is in good agreement with the experimental results (Hengehold et al. 1970) and the theoretical calculations (Wang et al. 2020). The first peak around 5 eV comes mainly from the transition between the Zn-3d and O-2p orbitals. The second peak energy around 10 eV comes from transition

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Fig. 4 The spin polarized density of states of pure ZnO and ZnO doping systems. The vertical lines at zero energy indicate the Fermi level.
between Zn-3d and O-2 s orbitals and the third one around 15 eV correspond to the transitions of Zn 4p state to O 2p state.

For Yb-doped ZnO systems, we find that the main dielectric peaks around 15 eV are significantly weakened and shift to lower energies with increasing Yb concentration, indicating that the range of absorption frequency narrows and the average optical transmittance increases. However, the peak around 10 eV increases with increasing Yb concentration due to the strong hybridization between the Zn-3d and O-2 s orbitals. Furthermore, compared to the dielectric function spectrum of pure ZnO, Yb-doped ZnO has a new peak around 1 eV which decreases and shifted to higher energies with increasing Yb concentration, It is the same behavior as that of iε2(ω) in (Jin et al. 2016; Wang et al. 2020) which, according to the study of state density and band structure, can be attributed to the transition of the Yb 4f states to the Zn 4 s states. Therefore, all the photons having a low energy can be absorbed by the doped ZnO, which relatively increased the absorption coefficient in the visible light region.

### 3.4.2 Absorption, reflectivity and transmittance

Figure 6 shows the calculated absorption and reflectivity of pure and Yb-doped ZnO with different concentrations of Yb. As can be seen from Fig. 6a, pure ZnO has an absorption peak around 350 nm, which gradually reduced and shifted toward the higher energy region with increasing Yb concentration. Meanwhile, in the visible light range of 500–800 nm, the absorption of Yb-doped ZnO was relatively enhanced compared to that of pure ZnO. This can be attributed to the Yb-4f defect induced near the conduction band minimum. Figure 6b shows the reflectivity of pure and Yb doped ZnO with deferent Yb concentrations; it can be observed that all pure and Yb doped ZnO structures have a low reflectivity which is less than 0.08 in the visible range. Compared to pure ZnO, the main peak about 400 nm is blue-shifted. A further decrease in the reflectivity spectrum was observed for Yb-doped ZnO in the visible region.

Figure 7 illustrates the measured transmittance of pure and Yb-doped ZnO with different concentrations of Yb. The average transmittance of pure ZnO is 88% in the visible region; the incorporation of Yb into ZnO increased transmittance in the range of 300–500 nm with a blue-shift to lower wavelengths, whereas it relatively decreased as a
result of the enhancement of absorption in the range of 500–800 compared with pure ZnO. The 4.17% Yb doped ZnO has a better transmittance than that of ZnO in the visible and UV regions.

### 3.4.3 Refractive index

Figure 8 shows the behavior of the refractive index (n) and the extinction coefficient (k) of ZnO and Yb-doped ZnO in the wavelength range of 200–800 nm; which is obtained from the relationship between the complex dielectric function and the complex refractive index. As shown in Fig. 8a, the static value of the refractive index n(0) at 0 eV of pure ZnO was 1.59 and the values of 1.92, 1.65 and 1.75 belonged to 4.17, 6.25 and 12.5% of Yb, respectively. Furthermore, a higher refractive index was obtained for pure ZnO with value 1.72 at 250 nm and 410 nm. As we can see, the static refractive index has changed considerably with increasing Yb. As the Yb concentration increases, the refractive index has changed considerably and shifts to higher energies with a maximum value of 1.76 obtained...
at 260 nm for 12.6% of Yb. However, in the visible range 380-780 nm, the refractive index decreased. This is consistent with the experimental findings of Ref. (López-Mena et al. 2020) which suggest that the decrease in the refractive index can be associated with variations in carrier concentration. Figure 8b shows the extinction coefficient (k) for pure ZnO and Yb-doped ZnO. A decrease in the extinction coefficient from 200 to 450 nm is observed. The opposite behavior is observed for wavelengths from 405 to 800 nm; this can be attributed to the impurity state induced by Yb doping, which is consistent with the calculation results of band structure and calculated transmittance observed in Figs. 3 and 7.

4 Conclusion

To sum up, we have investigated the changes of the electronic structure, magnetic and optical properties in ZnO induced by Yb doping using the GGA + U approximation based on DFT study. The obtained results showed an increase in the lattice parameters as the Yb concentration increased, which is in good agreement with the experimental findings. The band gap of ZnO increased with increasing Yb content and the Fermi level was shifted to the conduction band, revealing n-type characters with creating new defects related to the Yb-4f states. Both total and partial spin-polarized density of states calculations have been carried out to determine the value and the origin of induced magnetism in the non-magnetic ZnO system; the total magnetic moment increased and ZnO has a ferromagnetic property with an optimal concentration of 4.17% of Yb. Moreover, the absorption properties have been relatively increased in the visible range; a good transmittance was observed for 4.17% of Yb with a blue shift of the main peak compared to pure ZnO. Overall, the finding results indicate that Yb doped ZnO structures could be used as transparent conductive oxide. Hence, we believe that our findings will provide an effective means of understanding the doping effect in ZnO and will motivate further theoretical investigations.

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Declarations

Conflict of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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