First-Principle Studies on the Mechanical and Electronic Properties of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ ($x = 1\sim3$, $y = 1\sim2$, $z = 1\sim6$) Alloy under Pressure

Xiaoli Yuan 1,*, Weikang Li 1, Peng Wan 1 and Mi-An Xue 2,*

1 College of Science, Hohai University, Nanjing 210024, China; liweikang21@126.com (W.L.
2 Key Laboratory of Ministry of Education for Coastal Disaster and Protection, Hohai University, Nanjing 210024, China
* Correspondence: xlyuan@hhu.edu.cn (X.Y.); coexue@hhu.edu.cn (M.-A.X.)

Received: 20 August 2020; Accepted: 29 October 2020; Published: 5 November 2020

Abstract: The elastic and electronic properties of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ ($\text{AlNiZr}$, $\text{AlNi}_2\text{Zr}_5$, $\text{AlNi}_2\text{Zr}$, and $\text{Al}_3\text{Ni}_2\text{Zr}$) under pressure from 0 to 50 GPa have been investigated by using the density function theory (DFT) within the generalized gradient approximation (GGA). The elastic constants $C_{ij}$ (GPa), shear modulus $G$ (GPa), bulk modulus $B$ (GPa), Poisson’s ratio $\sigma$, Young’s modulus $E$ (GPa), and the ratio of $G/B$ have been studied under a pressure scale to 50 GPa. The relationship between Young’s modulus of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ is $\text{Al}_3\text{Ni}_2\text{Zr} > \text{AlNiZr} > \text{Al}_2\text{Ni}_2\text{Zr}_5 > \text{AlNi}_2\text{Zr}$, which indicates that the relationship between the stiffness of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ is $\text{Al}_3\text{Ni}_2\text{Zr} > \text{AlNiZr} > \text{Al}_2\text{Ni}_2\text{Zr}_5 > \text{AlNi}_2\text{Zr}$. The conditions are met at 30 and 50 GPa, respectively. What is more, the $G/B$ ratios for $\text{AlNiZr}$, $\text{AlNi}_2\text{Zr}$, $\text{Al}_2\text{Ni}_2\text{Zr}_5$, and $\text{Al}_3\text{Ni}_2\text{Zr}$ classify these materials as brittle under zero pressure, while with the increasing of the pressure the $G/B$ ratios of $\text{AlNiZr}$, $\text{AlNi}_2\text{Zr}$, $\text{Al}_2\text{Ni}_2\text{Zr}_5$, and $\text{Al}_3\text{Ni}_2\text{Zr}$ all become lower, which indicates that the pressure could enhance the brittle properties of these materials. Poisson’s ratio studies show that $\text{AlNiZr}$, $\text{AlNi}_2\text{Zr}$, and $\text{Al}_2\text{Ni}_2\text{Zr}_5$ are all a central force, while $\text{Al}_3\text{Ni}_2\text{Zr}$ is a non-central force pressure scale to 50 GPa. The energy band structure indicates that they are all metal. The relationship between the electrical conductivity of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ is $\text{Al}_3\text{Ni}_2\text{Zr} > \text{Al}_2\text{Ni}_2\text{Zr}_5 > \text{AlNiZr} > \text{AlNi}_2\text{Zr}$, which indicates that the relationship between the stiffness of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ is $\text{Al}_3\text{Ni}_2\text{Zr} > \text{Al}_2\text{Ni}_2\text{Zr}_5 > \text{AlNiZr} > \text{AlNi}_2\text{Zr}$. What is more, compared with $\text{Al}_3\text{Ni}_2\text{Zr}$, $\text{AlNiZr}$ has a smaller electron effective mass and larger atom delocalization. By exploring the elastic and electronic properties, they are all used as a superconducting material. However, $\text{Al}_3\text{Ni}_2\text{Zr}$ is the best of them when used as a superconducting material.

Keywords: $\text{Al}_x\text{Ni}_y\text{Zr}_z$; first-principle; electronic; mechanical properties; high pressure

1. Introduction

Ni-Al intermetallics are widely used as a novel high temperature structural material and aerospace material due to their high melting point, good oxidation resistance, and thermal conductivity. Shi et al. [1] by using the first-principle studied the electronic properties, elastic properties, structural properties, and the formation of Al-Ni intermetallics compounds. Wang et al. [2] studied the thermodynamic of Ni$_3$Al from the first-principle calculation. Chao et al. [3] by taking the first-principles, plane-wave method in combination with ultra-soft, pseudopotentials predict the crystal structures, lattice parameters, volumes, elastic constants, bulk moduli, and shear moduli of the binary NiAl. Yu et al. [4] investigated the self-diffusion in NiAl and Ni$_3$Al by the molecular dynamics (MD) with an analytical embedded atom and resolved this problem by incorporating Zr into Al-Ni to form the ternary Al-Ni-Zr [5,6].

However, its low ductility limited its application. It is an effectual way to enhance the oxidation resistance properties of the Ni-Al intermetallics by incorporating Zr into Al-Ni to form the ternary
Al-Ni-Zr [7,8], which arouses people’s interest in calculating the Zr-dropped Ni-Al intermetallics theoretically [9–11] and experimentally [12–15]. Wu et al. [16] studied the lattice misfit on the occupational behaviour and ductility properties with others, and found in energy analysis that the preferable site of Zr between Ni sublattice and Al sublattice will change under a different lattice misfit. Yang et al. [17] calculated the Gibbs energy of Zr-Al-Ni based on the quasi-regular solution model. Li et al. [18] conducted the research on the solid-liquid interfacial energy for Al-Ni-Zr alloys. However, there are limited reports on the elastic, structure, and electronic properties of Al-Ni-Zr intermetallics, including tetragonal Al$_5$NiZr$_2$, hexagonal Al$_3$NiZr$_6$, and AlNiZr, as well as the cubic AlNi$_2$Zr under pressure. What is more, the pressure dependence of the band gap and elastic properties for AlNi$_2$Zr is important to understand the effect of strain on the alloys and many practical applications [19].

Studying their properties can help us explore their potential and application value in the aerospace field. The first-principle calculation with the pseudopotential method based on DFT has grown up to be a standard tool for the calculation of the material modeling simulation [20–22]. Therefore, in this paper, the first-principle calculation with DFT and GGA is utilized to calculate the elastic and electronic properties.

2. Computational Method and Theory

The numerical calculations are done by using DFT with GGA and is performed by the exchange-correlation energy in the scheme of the Perdew-Burke-Ernzerhof (PBE). The pseudo atoms calculated are Al (3s$^2$ 3p$^1$), Ni (3d$^8$ 4s$^2$), and Zr (4s$^2$ 4p$^6$ 4d$^2$ 5s$^2$). In order to guarantee the accuracy of the calculation results, after repeated testing, we chose the cut-off energy of 600 eV for the wave function and charge density expansion. The k-point meshes of 4 × 4 × 8 for AlNiZr and Al$_2$NiZr$_6$, 6 × 6 × 6 for AlNi$_2$Zr, as well as 14 × 14 × 14 for Al$_3$Ni$_2$Zr, respectively were used to model the first Brillouin zone. The elastic properties were investigated by using the optimized stable structure. All calculations were done through the quantum mechanics software CASTEP [23].

AlNiZr and Al$_2$NiZr$_6$ belong to the hexagonal crystal system with space group P6-2m and they have five independent components $C_{11}$, $C_{33}$, $C_{44}$, $C_{12}$, and $C_{13}$, their stability equation was derived from [24]. AlNi$_2$Zr belongs to the cubic system with space group FM3-M and its elastic tensors $C_i$ have three independent components $C_{11}$, $C_{12}$, and $C_{44}$, whose equations are derived from [25]. Al$_5$Ni$_2$Zr with space group 14/MMM belongs to the tetragonal system and it has five independent components $C_{11}$, $C_{12}$, $C_{13}$, $C_{33}$, $C_{44}$, and $C_{66}$, whose stability equations are derived from [24]. The Voigt band and the Reuss band show the upper band and the lower band, respectively. Additionally, the Voigt-Reuss-Hill approximation means the arithmetic of the two bands [26]. $B$ and $G$ express the Bulk modulus and Shear modulus, respectively. $V$, $R$, and $H$ indicate the Voigt band, Reuss band, and Hill average, respectively. They are generated from the following equations.

For the cubic system [27]:

$$B_V = B_R = \frac{C_{11} + 2C_{12}}{3}$$

$$G_V = \frac{C_{11} - C_{12} + 3C_{44}}{5}$$

$$G_R = \frac{5C_{44}(C_{11} - C_{12})}{4C_{44} + (C_{11} - C_{12})}$$

The criterion of mechanical stability is [28–30], $P$ is the pressure.

$$C_{11} + 2C_{12} + P > 0, C_{44} - P > 0, C_{11} - C_{12} - 2P > 0$$

For the tetragonal system [31]:

$$B_R = [(2S_{11} + S_{33}) + 2(S_{12} + 2S_{13})]^{-1}$$
\[ \sigma_{ij} = \frac{1}{2} \sum_{k=1}^{3} C_{ij}^{k}\epsilon^{k} \]

For the hexagonal system [33]:

\[ B_R = \frac{(C_{11} + C_{12})C_{33} - 2C_{13}^2}{C_{11} + C_{12} + 2C_{33} - 4C_{13}} \]
\[ B_V = \frac{1}{9}[2(C_{11} + C_{12}) + C_{33} - 4C_{13}] \]
\[ \frac{G_R}{2} = \frac{5}{3}(C_{11} + C_{12})C_{33} - 2C_{13}^2 \frac{C_{44}C_{66}} {C_{44} + C_{66}} \]
\[ G_V = \frac{1}{30}(C_{11} + C_{12} + 2C_{33} - 4C_{13} + 12C_{44} + 12C_{66}) \]

The criterion of mechanical stability is [29]:

\[ C_{44} - P > 0; (C_{11} - P) > |C_{12} + P|; (C_{33} - P)(C_{11} - P + C_{12} + P) > 2(C_{13} + P) \]

The \( B_R, B_V, G_R, \) and \( G_V \) could be obtained by the calculation of Equations (1)–(11).

The average values under the Voigt and Reuss bounds can be expressed by the modulus of polycrystal, while under the Voigt-Reuss-Hill approximation [30,32,34]:

\[ B = \frac{1}{2}(B_V + B_R), \]
\[ G = \frac{1}{2}(G_V + G_R), \]

Young’s modulus (\( E \)) and Poisson’s ratio (\( \sigma \)) were obtained by these equations:

\[ E = \frac{9B_S}{3B_S + G}, \]
\[ \sigma = \frac{3B_S}{2(3B_S + G)}. \]

### 3. Results and Discussion

#### 3.1. The Elastic Properties under Pressure

To further understand the influence of the external pressure on the lattice parameters of Al\(_x\)Ni\(_y\)Zr\(_x\) alloys, the relative change equilibrium volume of Al\(_x\)Ni\(_y\)Zr\(_x\) in the range of 0–50 GPa is optimized and calculated with a step of 10 GPa, and the pressure and volume curves are drawn in Figure 1. The relative volume \( V/V_0 \) decreases with the increasing of the pressure, as shown in Figure 1.

The elastic constants were calculated using a linear fit of the stress-strain function. The relations calculated the elastic constant \( C_{ij} \) of Al\(_x\)Ni\(_y\)Zr\(_x\) with the pressure, as shown in Figure 2. The \( G \) (GPa), \( B \) (GPa), \( \sigma \), \( E \) (GPa), and the ratio of \( G/B \) at various pressures are listed in Table 1. Unfortunately, there are no experimental and theoretical elastic parameters to compare with Al\(_x\)Ni\(_y\)Zr\(_x\). First of all, the elastic constant of AlNiZr only meets the hexagonal mechanical stability criteria when \( P < 30 \) GPa.
When the pressure ranges from 0 to 50 GPa, the Al$_2$NiZr$_6$, AlNiZr$_2$, and Al$_3$Ni$_2$Zr are of mechanical stability and with no phase transformation until the pressure is up to 50 GPa. Therefore, in this paper, the elastic properties of Al$_2$NiZr$_6$, AlNiZr$_2$, and Al$_3$Ni$_2$Zr were studied and only the elastic and structural properties of AlNiZr under 30 GPa were studied. The data indicated that the relationship between Young’s moduli of Al$_x$Ni$_y$Zr$_z$ is Al$_5$Ni$_2$Zr > AlNiZr > Al$_2$NiZr$_6$ > AlNi$_2$Zr, which indicated that the relationship between the stiffness of Al$_x$Ni$_y$Zr$_z$ is Al$_5$Ni$_2$Zr > AlNiZr > Al$_2$NiZr$_6$ > AlNi$_2$Zr. The conditions are satisfied at 30 and 50 GPa, respectively.

![Figure 1](image1.png)

**Figure 1.** Calculated pressure dependence of the volume for Al$_x$Ni$_y$Zr$_z$.

![Figure 2](image2.png)

**Figure 2.** Calculated pressure dependence of the elastic constants $C_{ij}$ for Al$_x$Ni$_y$Zr$_z$ compounds: (a) AlNiZr, (b) AlNi$_2$Zr, (c) Al$_2$NiZr$_6$, (d) Al$_3$Ni$_2$Zr.
Table 1. The elastic constants $C_{ij}$, the Shear modulus $G$ (GPa), Bulk modulus $B$ (GPa), Poisson’s ratio $\sigma$, Young’s modulus $E$ (GPa), for (1) AlNiZr, (2) AlNi$_2$Zr, (3) Al$_2$NiZr$_6$, (4) Al$_3$Ni$_2$Zr at various pressures.

| Physical Quantities | G  | B  | E  | G/B | $\sigma$ |
|---------------------|----|----|----|-----|---------|
| P (GPa)             | 1  | 2  | 3  | 4   | 1  | 2  | 3  | 4   | 1  | 2  | 3  | 4   | 1  | 2  | 3  | 4   |
| 0                   | 64 | 45 | 54 | 87  | 140| 123| 106| 107 | 167| 120| 137| 205 | 0.46| 0.37| 0.51| 0.81 | 0.30| 0.34| 0.28| 0.18 |
| 10                  | 77 | 48 | 65 | 105 | 184| 133| 142| 134 | 203| 111| 170| 249 | 0.42| 0.32| 0.46| 0.78 | 0.32| 0.36| 0.29| 0.19 |
| 20                  | 88 | 56 | 74 | 124 | 220| 240| 165| 191 | 233| 155| 192| 306 | 0.40| 0.23| 0.45| 0.65 | 0.32| 0.39| 0.31| 0.23 |
| 30                  | 98 | 65 | 80 | 143 | 250| 274| 213| 212 | 261| 181| 213| 350 | 0.39| 0.24| 0.38| 0.68 | 0.33| 0.39| 0.33| 0.22 |
| 40                  | ...| 64 | 83 | 155 | 313| 240| 270 | 313| 179| 223| 392 | 0.20| 0.35| 0.57 | ...  | 0.41| 0.35| 0.26 |
| 50                  | ...| 65 | 88 | 168 | 343| 262| 303 | ... | 184| 236| 426 | 0.19| 0.30| 0.55 | ...  | 0.41| 0.35| 0.27 |
The variation relation of the elastic constants of Al$_x$Ni$_y$Zr$_z$ compounds with the pressure is shown in Figure 2. It can be learned from Figure 2 that a linear dependence in all the curves of these compounds are in the considered range of pressure. For Al$_2$NiZr$_6$, AlNi$_2$Zr, and AlNiZr, with the change of pressure $C_{11}$ changes the most compared with the other elastic moduli. $C_{44}$ changes the least with the pressure. As for Al$_2$Ni$_2$Zr, $C_{11}$ and $C_{33}$ are more sensitive to the change of pressure compared with $C_{44}$, $C_{66}$, $C_{12}$, and $C_{13}$.

From the relevant literature, it can be learned that G represents the plastic deformation resistance of the material and B represents the fracture resistance of the material [35]. Pugh proposed that the $G/B$ ratio is used to estimate the toughness of the corresponding materials [36]. According to his theory, a small $G/B$ value indicates the toughness of the corresponding material, while a larger $G/B$ value corresponds to brittleness. The critical separation value of ductile and brittle materials is around 1.75. If $G/B > 1.75$, the material is ductile, otherwise, the material is brittle [37].

The relations of Al$_x$Ni$_y$Zr$_z$ with the pressure and for AlNiZr, AlNi$_2$Zr, Al$_2$NiZr$_6$, and Al$_5$Ni$_2$Zr are presented in Figure 3 and Table 1. The $G/B$ ratios are respectively 0.457, 0.366, 0.372, 0.509, and 0.813 under zero pressure, classifying these materials as brittle. However, with the increasing of the pressure, the $G/B$ ratios of AlNiZr, AlNi$_2$Zr, Al$_2$NiZr$_6$, and Al$_5$Ni$_2$Zr all become lower which indicate that for these materials the pressure could enhance the brittle properties of the materials. The binding force of the atom can be shown by Poisson’s ratios. Poisson’s ratio of covalent materials is small (0.1), while the typical value of ionic materials is 0.25 [38]. Therefore, the ionic contribution of these compounds to the interatomic bond is dominant. The value of Poisson’s ratio represents the degree of directionality of covalent bonds. The values of 0.25 and 0.5 are the lower and upper limits of the central force solid, respectively [39]. As shown in Table 1, Poisson’s ratios of AlNiZr, AlNi$_2$Zr, and Al$_2$NiZr$_6$ are both bigger than 0.25 under the pressure from 0 to 50 GPa, similar to the AlNiZr under the pressure from 0 to 30 GPa, which shows that they are a central force. However, Poisson’s ratios of Al$_5$Ni$_2$Zr are less than 0.25 under the pressure from 0 to 30 GPa, which shows that it is a non-central force, while it is a central force when the pressure is higher than 30 GPa.

![Figure 3](image.png)

Figure 3. The dependence relations of $G/B$ values with pressures for Al$_x$Ni$_y$Zr$_z$ compounds.

### 3.2. The Electronic Properties

The calculated energy band structure of Al$_x$Ni$_y$Zr$_z$, (A) AlNiZr, (B) AlNi$_2$Zr, (C) Al$_2$NiZr$_6$, and (D) Al$_5$Ni$_2$Zr along with the high-symmetry directions in the Brillouin zone is shown in Figure 4.
which is in agreement with the increasing ground energy of $-9154.82$, $-9154.37$, $-9153.28$, $-9151.83$, $-9154.37$, and $-9148.31$ ev.

The Fermi level is the highest level at absolute zero. According to Pauli’s exclusion principle, a quantum state cannot hold two or more fermions. At absolute zero degrees, the electrons will fill the energy levels successively from low to high, except for the highest energy level, which will form the Fermi sea of electronic state. The plane of the Fermi sea is the Fermi level. The prerequisite for a good conductor is the case that the Fermi level intersects one or more energy bands. As shown in Figure 4, all the energy bands of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ pass through the Fermi surface, indicating that they are conductors. For $\text{AlNiZr}$ and $\text{Al}_2\text{NiZr}_6$ with the hexagonal system, the electrical conductivity of $\text{Al}_2\text{NiZr}_6$ is preferable to $\text{AlNiZr}$, due to more energy bands that pass through the Fermi surface for $\text{Al}_2\text{NiZr}_6$. For $\text{AlNiZr}$ and $\text{Al}_2\text{NiZr}_2$ with the cubic system, the electrical conductivity of $\text{Al}_3\text{Ni}_2\text{Zr}$ is preferable to $\text{AlNiZr}$. What is more, $\text{AlNi}_2\text{Zr}$ has a smaller electron effective mass and larger atomic non-localization than $\text{Al}_3\text{Ni}_2\text{Zr}$, similar to the bigger width of the band structure. $\text{Al}_3\text{Ni}_2\text{Zr}$ with the tetragonal system is expected to be a conductor. Figure 5 presents the total density of states of $\text{Al}_x\text{Ni}_y\text{Zr}_z$ under pressure from different pressures. It shows that the Fermi surface of $\text{AlNiZr}$ moves higher, which is consistent with the ground energy of $-8305.84$, $-8305.57$, $-8034.95$, and $-8034.12$ ev. The Fermi surface of $\text{Al}_2\text{NiZr}_6$ moves higher, which is in agreement with the increasing ground energy of $-9346.73$, $-9346.33$, $-9345.36$, $-9344.18$, $-9342.70$, and $-9341.07$ ev. The Fermi surface of $\text{AlNi}_2\text{Zr}$ moves higher, which is in agreement with the increasing ground energy of $-16,526.55$, $-16,526.17$, $-16,525.26$, $-16,524.01$, $-16,522.54$, and $-16,520.91$ ev. For $\text{Al}_3\text{Ni}_2\text{Zr}$, the Fermi surface moves higher, which is in agreement with the increasing ground energy of $-9154.82$, $-9154.37$, $-9153.28$, $-9151.83$, $-9154.37$, and $-9148.31$ ev.

Figure 4. Band structures of $\text{Al}_x\text{Ni}_y\text{Zr}_z$, (A) $\text{AlNiZr}$, (B) $\text{AlNi}_2\text{Zr}$, (C) $\text{Al}_3\text{Ni}_2\text{Zr}$, and (D) $\text{Al}_3\text{Ni}_2\text{Zr}$.
3.3. Superconducting Properties

By using the calculated elastic and electronic properties of the Al\textsubscript{x}Ni\textsubscript{y}Zr\textsubscript{z}, the possible superconducting properties were discussed. It can be learned from the simplified theory of superconductivity that the material needs to meet three conditions to become a superconductor. Firstly, the crystal’s atoms are lighter. Secondly, the coefficient of elasticity of the crystal is as large as possible and the crystal is tough enough. Thirdly, the material should be tantamount to the lower effective Fermi level. Al\textsubscript{x}Ni\textsubscript{y}Zr\textsubscript{z} meets the first condition. What is more, Al and Zr are among the 28 superconducting elements. By observing the result of the elastic properties, Al\textsubscript{5}Ni\textsubscript{2}Zr is tougher than Al\textsubscript{2}NiZr\textsubscript{6}, AlNiZr, and AlNi\textsubscript{2}Zr. From the third point of view, both of them are the metal system. In conclusion, they are all liable to be used as a superconducting material. However, Al\textsubscript{5}Ni\textsubscript{2}Zr is the best of them when used as a superconducting material [26].

3.4. Difference Charge Density

The difference charge density maps are shown in Figure 6, which can clearly reflect the bonding between atoms of Al\textsubscript{x}Ni\textsubscript{y}Zr\textsubscript{z}. By observing the plots for AlNiZr, the charge density between Ni and Ni are smaller than that between Al and Ni, Zr, and Ni, which shows that the effect between Ni and Ni is stronger. For Al\textsubscript{2}NiZr\textsubscript{6}, the charge density between Al and Zr is bigger than that of Al and Al. What is more, for AlNiZr and Al\textsubscript{2}NiZr\textsubscript{6} with the same system, the interatomic interaction of AlNiZr is greater.
than that of Al$_5$Ni$_2$Zr. For AlNi$_2$Zr, there is a larger charge density between Al and Ni, Zr, and Ni than Zr and Al, which means that the effect between Al and Ni, Zr, and Ni is stronger than Zr and Al. For Al$_5$Ni$_2$Zr, the charge density between Al and Ni is larger than that of Al and Zr, Al, and Al, which means that there is a stronger effect between Al and Ni than Al and Zr, Al, and Al. In addition, the interatomic interaction of Al$_5$Ni$_2$Z is greater than that of Al$_2$NiZr$_6$.

![Figure 6. Difference charge density maps of Al$_x$Ni$_y$Zrz.](image)

(A) AlNiZr, (B) Al$_2$NiZr$_6$, (C) AlNi$_2$Zr, (D) Al$_5$Ni$_2$Zr.

4. Conclusions

The first-principle study on elastic and electronic properties of hexagonal AlNiZr and Al$_2$NiZr$_6$, cubic AlNi$_2$Zr, as well as tetragonal Al$_5$Ni$_2$Zr under pressure from 0 to 50 GPa have been calculated by using the plane-wave, ultra-soft, pseudopotentials technique within the generalized gradient approximation (GGA). The elastic constants, volume, and density of states change under pressure are analyzed. The volume decreases and the ground energy of material increases with the increase of the pressure, which is in agreement with the Fermi surface moves to be higher. The elastic constants $C_{ij}$, Shear modulus $G$ (GPa), Bulk modulus $B$ (GPa), Poisson’s ratio $\sigma$, Young’s modulus $E$ (GPa), and the ratio of $G/B$ have been calculated under pressure from 0 to 50 GPa. Young’s modulus indicated that the relationship between the stiffness of Al$_x$Ni$_y$Zrz is Al$_5$Ni$_2$Zr > AlNiZr > Al$_2$NiZr$_6$ > AlNi$_2$Zr. The conditions are met at 30 and 50 GPa, respectively. For Poisson’s ratio, AlNiZr from 0 to 30 GPa and AlNi$_2$Zr as well as Al$_5$NiZr$_6$ from 0 to 50 GPa are all a central force, while Al$_5$Ni$_2$Zr is a non-central force from 0 to 50 GPa. The $C/B$ ratios for AlNiZr, AlNi$_2$Zr, Al$_2$NiZr$_6$, and Al$_5$Ni$_2$Zr classify these materials as brittle under zero pressure, while with the increase in pressure the $C/B$ ratios of AlNiZr, AlNi$_2$Zr, Al$_2$NiZr$_6$, and Al$_5$Ni$_2$Zr all become lower, which indicate that the pressure could enhance the brittle properties of these materials.

By observing the density difference charge, the effect between Ni and Ni is stronger for AlNiZr. The effect between Al and Zr is bigger than that of Al and Al for Al$_2$NiZr$_6$. What is more, for AlNiZr and Al$_2$NiZr$_6$, with the same system, the interatomic interaction of AlNiZr is greater than that of Al$_2$NiZr$_6$. For AlNi$_2$Zr, the effect between Al and Ni, Zr, and Ni is stronger than Zr and Al. For Al$_5$Ni$_2$Zr, there is a stronger effect between Al and Ni than Al and Zr, Al, and Al. In addition, the interatomic interaction of Al$_5$Ni$_2$Z is greater than that of Al$_2$NiZr$_6$.

The relationship between the electrical conductivity of Al$_x$Ni$_y$Zrz is Al$_2$NiZr$_6$ > Al$_5$Ni$_2$Zr > AlNi$_2$Zr > AlNiZr. What is more, AlNiZr has a smaller electron effective mass and larger atomic non-localization than Al$_5$Ni$_2$Zr. By studying the elastic and electronic properties, they all have the
potential be used as a superconducting material. However, Al$_3$Ni$_2$Zr is the best of them when used as a superconducting material.

**Author Contributions:** Conceptualization, X.Y.; funding acquisition, X.Y. and M.-A.X.; investigation, W.L. and P.W.; methodology, W.L. and P.W.; writing—original draft, W.L.; writing—review & editing, X.Y. and M.-A.X. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research received no external funding.

**Acknowledgments:** This research is supported by the Fundamental Research Funds for the Central Universities (No. 2019B19314) and the National Natural Science Foundation of China (Nos. 51979097 and 51609076).

**Conflicts of Interest:** The authors declare no conflict of interest.

**References**

1. Shi, D.; Wen, B.; Melnik, R. First-principles studies of Al-Ni intermetallic compounds. *J. Solid State Chem.* 2009, 182, 2664–2669. [CrossRef]
2. Wang, Y.; Liu, Z.K.; Chen, L.Q. Thermodynamic properties of Al, Ni, NiAl and Ni$_3$Al from first-principles calculations. *Acta Mater.* 2004, 52, 2665–2671.
3. Xu, C.; Chen, D. Computer Simulation: A Tool for Researching the Basic Physical Properties of NiAl and Ni$_3$Al. *Adv. Mat. Res.* 2014, 989–994, 216–219.
4. Yu, S.; Wang, C.Y.; Yu, T. Self-diffusion in the intermetallic compounds NiAl and Ni$_3$Al: An embedded atom method study. *Phys. B* 2007, 396, 138–144. [CrossRef]
5. George, E.P.; Liu, C.T.; Pope, D.P. Environmental embrittlement: The major cause of room-temperature brittleness in polycrystalline Ni$_3$Al. *Scripta Metall. Mater.* 1992, 27, 365–370. [CrossRef]
6. George, E.P.; Liu, C.T.; Pope, D.P. Intrinsic ductility and environmental embrittlement of binary Ni$_3$Al. *Scripta Metall. Mater.* 1993, 28, 857–862. [CrossRef]
7. Chang, T.T.; Pan, Y.C.; Chuang, T.H. The oxidation behavior of Ni$_3$AlZr alloys with various zirconium contents. *J. Alloys Compd.* 1996, 243, 126–132. [CrossRef]
8. Lee, D.B.; Santella, M.L. High temperature oxidation of Ni$_3$Al alloy containing Cr, Zr, Mo, and B. *Mater. Sci. Eng. A* 2004, 374, 217–223. [CrossRef]
9. Kim, D.E.; Shang, S.L.; Liu, Z.K. Effects of alloying elements on elastic properties of Ni by first-principles calculations. *Comput. Mater. Sci.* 2009, 47, 254–260. [CrossRef]
10. Kim, D.E.; Shang, S.L.; Liu, Z.K. Effects of alloying elements on thermal expansions of γ-Ni and γ′-Ni$_3$Al by first-principles calculations. *Acta Mater.* 2012, 60, 1846–1856. [CrossRef]
11. Zhang, C.M.; Guo, Y.Q.; Yuan, Z.P. Effect of octadeca carbon fatty acids on microbial fermentation, methanogenesis and microbial flora in vitro. *Anim. Feed Sci. Technol.* 2008, 146, 259–269. [CrossRef]
12. Ball, J.; Zeumer, B.; Gottstein, G. Large strain deformation of Ni$_3$Al+B: IV. The effect of Zr and Fe additions. *Intermetallics* 1995, 3, 209–219. [CrossRef]
13. Li, Y.F.; Guo, J.T.; Zhou, L.Z. Effect of recrystallization on room-temperature mechanical properties of Zr-doped Ni$_3$Al alloy. *Mater. Lett.* 2004, 58, 1853–1856. [CrossRef]
14. Motejadded, H.B.; Soltanieh, M.; Rastegari, S. Dissolution Mechanism of a Zr Rich Structure in a Ni$_3$Al Base Alloy. *J. Mater. Sci. Technol.* 2011, 27, 885–892. [CrossRef]
15. Li, Y.F.; Guo, J.T.; Shen, Y.F. Influence of recrystallization and environment on tensile behavior of cold-rolled Ni$_3$Al(Zr) alloys. *Trans. Nonferrous Met. Soc. China* 2006, 16, 368–375. [CrossRef]
16. Wu, Y.; Zhang, W.; Guo, J.; Hou, J.; Li, X.; Huang, R.; Ma, X.; Zhang, Q. The First-principles Study on the Occupation Behavior and the Ductility Mechanism of Zr in Ni-Ni$_3$Al System with Lattice Misfit. *J. Mater. Sci. Technol.* 2014, 30, 517–522. [CrossRef]
17. Li, H.Q.; Yang, Y.S.; Tong, W.H. Gibbs energy calculation of liquid Zr-Al-Ni and Zr-Al-Cu-Ni alloys with clusters. *J. Mater. Sci.* 2007, 42, 4060–4065. [CrossRef]
18. Li, H.Q.; Yang, Y.S.; Tong, W.H. Calculation of the solid-liquid interfacial energy for Zr-Ni-Al and Zr-Ni-Al-Cu alloys based on the non-structural approach. *Model. Simul. Mater. Sci. Eng.* 2006, 14, 1095–1103. [CrossRef]
19. Liu, Q.J.; Liu, Z.T.; Feng, L.P. First-Principles Calculations of Structural, Elastic and Electronic Properties of Tetragonal HfO$_2$ under Pressure. *Commun. Theor. Phys.* 2011, 56, 779–784. [CrossRef]
20. Gonze, X.; Beuken, J.M.; Caracas, R. First-principles computation of material properties: The ABINIT software project. Comp. Mater. Sci. 2002, 25, 478–492. [CrossRef]
21. Shang, S.L.; Wang, Y.; Kim, D.E. First-principles thermodynamics from phonon and Debye model: Application to Ni and NiAl. Comp. Mater. Sci. 2010, 47, 1040–1048. [CrossRef]
22. Wang, B.T.; Shi, H.; Li, W.D. First-principles study of ground-state properties and high pressure behavior of ThO$_2$. J. Nucl. Mater. 2010, 399, 181–188. [CrossRef]
23. Payne, M.C.; Arias, T.A.; Joannopoulos, J.D. Iterative minimization techniques for ab initio total-energy calculations: Molecular dynamics and conjugate gradients. Rev. Mod. Phys. 1992, 64, 1045–1097. [CrossRef]
24. Mouhat, F.; Coudert, F.X. Necessary and sufficient elastic stability conditions in various crystal systems. Phys. Rev. B 2014, 90, 224104. [CrossRef]
25. Haines, J.; Leger, J.M.; Bocquillon, G. Synthesis and design of superhard materials. Annu. Rev. Mater. Sci. 2001, 31, 1–23. [CrossRef]
26. Ji, Y.Q.; Yuan, X.L. Elastic Properties and Electronic Properties of M$_x$N$_y$ (M = Ti, Zr) from First Principles Calculations. Materials 2018, 11, 1–23. [CrossRef]
27. Zha, C.S.; Mao, H.K.; Hemley, R.J. Elasticity of MgO and a primary pressure scale to 55 GPa. Proc. Natl. Acad. Sci. USA 2000, 97, 13494–13499. [CrossRef]
28. Sinko, G.V.; Smirnov, N.A. Ab initio calculations of elastic constants and thermodynamic properties of bcc, fcc, and hcp Al crystals under pressure. J. Phys. Condens. Matter 2002, 14, 6989–7005.
29. Yip, S.; Li, J.; Tang, M.J.; Wang, J.H. Mechanistic aspects and atomic-level consequences of elastic instabilities in homogeneous crystals. Mater. Sci. Eng. A 2001, 317, 236–240. [CrossRef]
30. Xue, M.; Yuan, X.; Zhong, C.; Wan, P. First principles calculations on elastic, thermodynamic and electronic properties of Co$_2$Zr and Co$_2$Ti at high temperature and pressure. Appl. Sci. 2020, 10, 2097. [CrossRef]
31. Reuss, A. Berechnung der Fließgrenze von Mischkristallen auf Grund der Plastizitätsbedingung für Einkristalle. J. Appl. Math. Mech. 1929, 9, 49–58. [CrossRef]
32. Hill, R. The Elastic Behaviour of a Crystalline Aggregate. Proc. Phys. Soc. A 1952, 65, 349–354. [CrossRef]
33. Li, X.Z.; Liu, L.; Ding, C.L. First-principles investigations on mechanical stability and elastic properties of hexagonal tungsten dinitride under pressure. Mater. Chem. Phys. 2011, 130, 14–19. [CrossRef]
34. Born, M. On the stability of crystal lattices. I. Math. Proc. Camb. Philos. Soc. 1940, 36, 160–172. [CrossRef]
35. Ghebouli, B.; Fatmi, M.; Ghebouli, M.A.; Choutri, H.; Louail, L.; Chihi, T.; Bouhemadou, A.; Bin-Omran, S. Theoretical study of the structural, elastic, electronic and optical properties of XCaF$_3$ (X = K and Rb). Solid State Sci. 2015, 43, 9–14. [CrossRef]
36. Chen, X.Q.; Niu, H.; Li, D. Modeling hardness of polycrystalline materials and bulk metallic glasses. Intermetallics 2011, 19, 1275–1281. [CrossRef]
37. Haddadi, K.; Bouhemadou, A.; Louail, L. Structural and elastic properties under pressure effect of the cubic antiperovskite compounds ANCa$_3$ (A = P, As, Sb, and Bi). Phys. Lett. A 2009, 373, 1777–1781. [CrossRef]
38. Bannikov, V.V.; Shein, I.R.; Ivanovskii, A.L. Ab initio Predictions of Stability and Electronic Properties of Cubic Rhodium Carbides RhCx as Dependent on Carbon Content. Phys. Stat. Solidi Rapid Res. Lett. 2009, 3, 218–220. [CrossRef]
39. Yuan, X.L.; Xue, M.A.; Chen, W.; An, T.Q.; Cheng, Y. Investigations on the structural, elastic and electronic properties of the orthorhombic Zirconium–Nickel alloy under different pressure. Comput. Mater. Sci. 2012, 65, 127–132. [CrossRef]

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.