Data Article

Data on electrical properties of nickel modified potassium polytitanates compacted powders
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ARTICLE INFO

Article history:
Received 5 May 2015
Accepted 18 May 2015
Available online 28 May 2015

Keywords:
Modified potassium titanates
Permittivity
Conductivity
Dielectric losses

ABSTRACT

Potassium polytitanates are new promising type of ferroelectric ceramic materials with high ionic conductivity, highly polarizable structure and extremely high permittivity. Its structure is formed by [TiO₆] octahedral units to layers with mobile potassium and hydroxonium ions in-between. The treatment in solutions containing nickel ions allows forming heterostructured materials which consist of potassium polytitanate particles intercalated by Ni²⁺ ions and/or decorated by nickel oxides NiOₓ. This modification route is fully dependant on solution pH, i.e. in acidic solutions the intercalation process prevails, in alkaline solutions potassium polytitanate is mostly decorated by the oxides. Therefore, electronic structure and electrical properties can be regulated depending on modification conditions, pH and ions concentration. Here we report the data on electric properties of potassium titanate modified in nickel sulfate solutions at different pH.

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http://dx.doi.org/10.1016/j.dib.2015.05.010
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1. Experimental design, materials and methods

The obtained parent and modified potassium polytitanate (PPT) powders were studied by XRD using ARL XTRA Thermo Scientific what confirmed its (X-ray) amorphous structure. The amorphous structure of the titanates remains unchanged up to 500–600 °C [1]. Chemical composition was confirmed by Energy-Dispersive X-ray Fluorescence Spectrometer BRA-135F (Table 1).

Water contents in the investigated powders were measured using thermo-gravimetric method (STD Q600 TA Instruments). According to the obtained data and our previous results physically adsorbed water is desorbed from PPT powders at 100–220 °C, whereas structural water (located in the interlaminar space of PPT particles) is being removed at 350–500 °C [1].

Impedance spectroscopy was conducted in the frequency range $10^{-2}$–$10^{6}$ Hz applying periodic signal 50 mV (impedance-meter Novocontrol Alpha-AN). Electrical contacts were realized by claying metal wires with conductive paste Contactol-K13 to the round sides of the compacts.

### Table 1

| Conditions | X-ray fluorescence analysis data | TGA data |
|------------|---------------------------------|----------|
|            | $K_2O$ | TiO$_2$ | NiO | Al$_2$O$_3$ | SiO$_2$ | Physically adsorbed H$_2$O (25 °C < T < 220 °C) | Structural H$_2$O (350 °C < T < 500 °C) |
| (pH=3.0)   | 0.7    | 95.9    | 1.2 | 1.2         | 1.0     | 8.3                                | 4.2            |
| (pH=6.5)   | 1.8    | 82.2    | 15.2| 0.2         | 0.6     | 7.6                                | 2.5            |
| (pH=9.0)   | 4.7    | 73.4    | 20.5| 0.8         | 0.6     | 4.8                                | 1.8            |
| Parent PPT powder (pH=10.1) | 19.5 | 79.8    | –   | –           | 0.7     | 7.8                                | 3.1            |
The rest surface was isolated by epoxy compound Pentelepst 1143-A. Obtained data was used to calculate conductivity $\sigma$, dielectric dissipation factor, $\tan(\delta)$, and complex permittivity $\varepsilon^*$ [2].

2. Material preparation

The PPT powder was produced by the treatment of powdered TiO$_2$ (anatase, Aldrich 99%+, average size of particles ca. 7 $\mu$m) in the molten mixture of KNO$_3$ and KOH at 500 $^\circ$C for 2 h according to [1]. The TiO$_2$:KOH:KNO$_3$ weight ratio was 1:1:8. The synthesized potassium polytitanate was carefully washed with distilled water three times (1 dm$^3$ of H$_2$O per 100 g of the product for each washing), filtrated (Whatman paper No 40) and dried at 60 $^\circ$C in an oven to obtain the parent (X-ray) amorphous PPT powder characterized with the TiO$_2$:K$_2$O molar ratio 4:1.

The parent PPT powder was modified according to the following route [3,4]: 30 g of powder was put in a glass which then was filled up with 200 ml of 0.001 M nickel sulfate solution. pH of the obtained dispersion was adjusted by additions of 10% H$_2$SO$_4$ or 10% KOH to acidic (pH = 3.0), neutral (6.5) or alkaline (9.0). The obtained suspension was stirred 4 h and then the powder was separated and dried at 60 $^\circ$C.

![Fig. 1. Bode conductivity plots for compacted parent PPT (1) and PPT modified in solution containing nickel ions at pH=3.0 (2), 6.5 (3) or 9.0 (4) before (A) and after the thermal treatment (B).](image-url)
Modified PPT powders were compacted to density 2.2 g/cm$^3$ in disks of 8 mm of diameter and 2.0 ± 0.1 mm of thickness. Electric properties were studied with the parent compacts and with similar specimens obtained after the thermal treatment of parent compacts in the oven at 160 °C for 6 h for total desorption of the physically adsorbed water. All the measurements were conducted at room temperature.

3. Chemical composition data

The powders modified in acidic solution show a relatively low content of NiO. Modification in alkaline solution results in high amount of both NiO and K$_2$O in the PPT powder. Particularly, K$_2$O content is high, as a consequence of a low ion-exchange rate. The powders obtained in neutral solution (pH = 6.5, close to isoelectrical point for nickel sulfate aqueous solution, 0.001 M) show moderate values for NiO and K$_2$O which are higher than in acidic solution but lower than in alkaline media [3].

![Fig. 2. Bode tan(δ) plots for compacted parent PPT (1) and PPT modified in solution containing nickel ions at pH = 3.0 (2), 6.5 (3) and 9.0 (4) before (A) and after the thermal treatment (B).](image-url)
4. Data on electrical properties of the PPT compacts modified in nickel sulfate solution

Bode plots for modified PPT compacts before and after the thermal treatment are presented in Figs. 1–3.

The conductivity data show a linear change at low frequencies (Fig. 1). By extrapolating it to the frequency $10^{-2}$ Hz one might assess dc conductivity, $\sigma_{dc}$, in dependence on composition of particular compact before or after the thermal treatment. Compacts before the thermal treatment (containing physically adsorbed water) show dc-conductivity similar to the one for the parent PPT ($\sim 5 \cdot 10^{-9}$ Sm/cm) [5]. However, compacts modified at pH = 6.5 show lower conductivity $\sigma_{dc} \sim 10^{-9}$ Sm/cm. After removal of adsorbed water from the PPT/Ni modified at pH 6.5 its conductivity $\sigma_{dc}$ remains the same $10^{-9}$ Sm/cm. Though, for PPT/Ni (pH = 3.0) the conductivity $\sigma_{dc}$ drops to $\sim 10^{-10}$ Sm/cm.

The highest ac conductivity, $\sigma_{ac}$, at $\sim 10^6$ Hz is found for the parent PPT powder compacts and the compacts of PPT modified in alkaline solution ($\sigma_{ac} \sim 10^{-5}$ Sm/cm), i.e. with relatively high $\text{K}_2\text{O}$ content. After the thermal treatment and consequent removal of the adsorbed water conductivity shows a change of about an order ($\sim 10^{-6}$ Sm/cm) for the modified PPT compacts and the drop is 1.5 orders ($\sim 5 \cdot 10^{-7}$ Sm/cm) for the parent PPT compacts.

Fig. 3. Bode complex permittivity plots for compacted parent PPT (1) and PPT modified in solution containing nickel ions at pH=3.0 (2), 6.5 (3) and 9.0 (4) before (A) and after the thermal treatment (B).
Data on $\tan(\delta)$ shows a presence of several relaxation peaks. The peaks might be attributed to two relaxation processes with relaxation times of $6\text{–}60$ s and $0.1\text{–}1$ s respectively. The first peak is presumably due to proton transfer along grain boundaries of the compacted particles for pure PPT or for their modified specimens, which are covered with physically absorbed water. This correlates with the results for $\tan(\delta)$ obtained for the samples before the thermal treatment. The second peak, appeared at lower frequencies, should be due to proton transfer via structural water localized in the interlayer space of PPT particles. The potassium polytitanate compacts modified at low pH ($\text{pH}=3.0$) possess the highest values of $\tan(\delta)$ in the region $1\text{–}10^5$ Hz while a product obtained by modification in neutral and alkaline solutions and, therefore, less hydrated, shows the lowest $\tan(\delta)$ values. Correspondingly, conductivity of the compacts PPT/Ni ($\text{pH}=3.0$) is maximal at lower frequencies. Bode plots of dielectric dissipation factor for compacted PPT powders with high $K_2O$ content (parent PPT and PPT/Ni (pH=9.0)) (Fig. 2A) shows characteristic peaks at $\sim 10^3$ Hz. At the same time, for the compacts with low $K_2O$ content PPT/Ni (pH=6.5 and pH=3.0) the peak is at $\sim 10^2$ Hz. A small maximum is also observed at 0.01 Hz as well for PPT compact prepared from parent powder.

After the thermal treatment, i.e. removal of adsorbed water (Fig. 2B) maxima of the peaks of $\tan(\delta)$ for all the compacts are shifted to low frequency region ($\sim 0.5$ Hz), with maximum intensity referred to a peak for PPT modified in acidic solution. Two peaks are observed for PPT/Ni (pH=9.0) at $10^{-1}$ Hz and 10 Hz.

All the investigated compacts can be characterized by a high permittivity ($\sim 10^6$) at low frequencies and permittivity $\sim 10^2$ in high frequencies region. It is worth highlighting that the permittivity of PPT compacts modified in the neutral solution is lower comparing to the other compacts and equals $4\cdot10^5$ at low frequencies. After removal of physically adsorbed water it decreases to $(5\text{–}7)\cdot10^6$ following the same trend for the studied compacts. At the same time for PPT/Ni obtained in acidic solution, permittivity remains at value of $5\cdot10^5$. At high frequencies all modified PPT compacts show higher permittivity values comparing to parent PPT; the highest permittivity (95) is ascribed for PPT/Ni compact modified in acidic solution.

The best polarization properties (the highest permittivity) are confirmed for the compacts obtained from PPT powders by their treatment in aqueous solutions of nickel sulfate at pH=6.5. This product is characterized with complicated structure consisting of PPT particles partially intercalated with Ni$^{2+}$ ions and decorated by NiO nanoparticles.

This data shows that modification of PPT powders in aqueous solutions of nickel sulfate allows obtaining the materials with high permittivity in low-frequency region as well as with low ionic and electronic conductivity. The compacts produced with the PPT/Ni (pH=3.0) powder thermally treated at 160 °C are characterized with high value of permittivity which remains up to the temperature of 350–400 °C while content of structural water in the composition of this powder does not change significantly.

**Acknowledgment**

This work is performed in frame of the Federal Target Program of Russian Ministry of Education and Science “Research and Development of Scientific–Technological Complex of Russia in 2014–2020 on directions of high priority” (agreement 14.574.21.0005 from 17.06.2014).

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