New Method for the Synthesis of 2D Vanadium Nitride (MXene) and Its Application as a Supercapacitor Electrode

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1. INTRODUCTION

Because of the imminent global energy crisis, research is being focused on developing efficient energy conversion and storage systems. Supercapacitors have been gaining much importance because of their high power densities and long cycle life.1,2 Supercapacitors bridge the existing gap between batteries and capacitors, which exhibit characteristics of high energy density and high power density, respectively.3 The performance of these supercapacitors can be enhanced by constructing them with efficient electrode materials.4 In the recent years, two-dimensional (2D) materials have been gaining limelight with the excellent properties exhibited by graphene. Extensive studies have been carried out on 2D materials such as transition metal dichalcogenides (TMDs), hexagonal boron nitride, metal–organic frameworks, and the newly discovered MXenes as electrodes for supercapacitors. MXenes, discovered by the Gogotsi and Barsoum group in 2011, belong to the family of 2D materials, which includes transition metal nitrides, carbides, and carbonitrides.5 MXenes are prepared by selective etching of A elements from 3D MAX phases, which has a general formula M_{n+1}AX_n (n = 1, 2, and 3), where M, A, and X represent the d-block transition metals, group 13 or 14 elements, and either or both C and N atoms, respectively.5 The new material MXene was named because of the loss of the A element from the parent MAX phase and its structure analogous to graphene. MAX phases are layered hexagonal structures with P63/mmc symmetry, wherein the X atoms are sandwiched between the M layers and A atoms are present in the interleaved spaces of M_{n+1}X_n layers.7 The bond between M–X exhibits strong mixed covalent/ionic/metallic nature, whereas the bond between M–A is metallic in nature. Thus, the bonds in the MAX phases are highly strong to be broken but using the phenomenon of relative bond energies between M–A and M–X bonds, the “A” atoms can be selectively etched from the MAX phases by various chemical means without...
disrupting the M–X bonds.\(^8\) After the etching process, chemically stable and closely packed MXenes with formula \(M_nX_T\) will be obtained, where \(T\) refers to the functional groups (–OH, –O, and –F) attached to the surface of the metal atoms after the etching procedure.\(^9\)

Etching of “A” element from the MAX phase is the crucial step in the synthesis of MXene. The M–A bonds are weaker when compared to that of M–X bonds and the “A” element in the MAX phase can be etched through various methods reported in the literature. HF is the widely used etchant for carbide MXenes and its concentration, exposure time, and temperature are all crucial parameters in the process of etching.\(^10\) Milder etchants such as NaHF\(_2\), KHF\(_2\), or NH\(_2\)HF\(_2\) and a mixture of acid (H\(_2\)SO\(_4\)/HCl) and fluoride salt (NaF, KF, and LiF) have also been reported as effective etchants.\(^11\) In the case of nitride MXenes, HF is incapable of etching “A” from their MAX phases because of its lower cohesive energies, which enables it to dissolve in HF. Urbankowski et al.\(^12\) reported the synthesis of Ti\(_3\)N\(_3\) using a eutectic mixture of molten salts as etching agents. The NaF–HCl mixture was used in the etching process by Soundiraraju and George to obtain Ti\(_3\)N.\(^13\) In the current work, the LiF–HCl mixture was used as an etchant to obtain V\(_2\)N from V\(_2\)AlN. To obtain few-layered MXenes from the multilayered MXenes, usually, tetrabutylammonium hydroxide (TBAOH) and dimethyl sulfoxide (DMSO) are used as the delaminating agents.\(^12,13\)

According to Barsoum and Radovic, more than 60 MAX phases exist till date but only a few MXenes from those are reported so far.\(^14\) More many MXenes are expected to be produced from their parent MAX phases in the future, thus providing a lot of scope in this field of research. Carbide- and carbonitride-based MXenes such as Ti\(_3\)C,\(^15\) V\(_2\)C,\(^16\) Nb\(_2\)C,\(^17\) Mo\(_2\)C,\(^18\) Ti\(_3\)C\(_2\),\(^19\) Zr\(_2\)C\(_2\),\(^19\) Nb\(_2\)C\(_2\),\(^20\) Ta\(_2\)C\(_2\),\(^19\) Hf\(_2\)C\(_2\),\(^21\) Ti\(_3\)Nb,\(_2\),\(^22\) Mo\(_2\)TiC\(_2\),\(^22\) (Ti\(_{5.8}\)Nb\(_{2.5}\))\(_2\)C\(_3\),\(^22\) (V\(_{5.8}\)Cr\(_{2.5}\))\(_2\)C\(_3\),\(^22\) Mo\(_2\)ScC\(_2\),\(_23\) Cr\(_2\)TiC\(_2\),\(_22\) Mo\(_2\)Ti\(_2\)C\(_3\),\(_22\) (Nb\(_{0.8}\)Ti\(_{0.2}\))\(_4\)C\(_3\),\(_22\) (Nb\(_{0.8}\)Zr\(_{0.2}\))\(_4\)C\(_3\),\(_22\) and Ti\(_3\)CN are some of the reported MXenes. Nitride-based MXenes have been rarely explored and only a few of them have been reported recently, viz., Ti\(_3\)N,\(_22,23\) and Ti\(_2\)N.\(_24,26\) Nitride MXenes obtained by the ammonization of carbide MXenes have also been reported.\(^27\)

MXenes have started gaining a lot of attention since its discovery and have been explored in various fields.\(^28\) Xie et al. reported the use of MXenes as a promising electrode material for rechargeable non-lithium-ion batteries.\(^29\) 2D Zr\(_2\)CO\(_3\) and H\(_2\)CO\(_3\) were proved as an efficient catalyst for high-efficiency photocatalytic water splitting, thereby enabling the production of hydrogen fuel.\(^30\) The antibacterial properties of Ti\(_3\)C\(_2\) were explored against Escherichia coli and Bacillus subtilis, where it exhibited a higher antibacterial efficiency compared to graphene oxide.\(^31\) A Ti\(_3\)C\(_2\) MXene-FET biosensor developed by the Xu and group was successful in monitoring the activity in hippocampal neurons with high sensitivity through a label-free detection of dopamine.\(^32\) The effectiveness of Ti\(_3\)C\(_2\) in the absorbing and shielding of electromagnetic interferences in the X-band was reported by Han et al.\(^33\) MXene has found its importance extensively in the field of energy storage, particularly, as an electrode material for supercapacitors. Yang et al. reported the use of Ti\(_3\)C\(_2\)/graphene hybrid fibers spun using wet-spinning assembly as an electrode material for supercapacitors, exhibiting a superior volumetric capacitance of 86.4 F cm\(^{-3}\). A flexible supercapacitor electrode was fabricated using surfactant-free Ti\(_3\)C\(_2\)T\(_x\) flakes loaded on a carbonized silk cloth, which exhibited excellent cyclability and flexibility.\(^34\) Thus, MXenes find applications in almost all the areas of science and technology, leaving a space of interest to probe more into the synthesis of MXenes. There are numerous reports in the literature of using carbide-based MXene as supercapacitor electrodes but only few exist for a nitride-based MXene, which may be because of the difficulty in its synthesis.\(^25,26,36\)

In this work, we report a unique and facile protocol to synthesize V\(_2\)NT\(_x\) MXene and its application as a supercapacitor electrode. The V\(_2\)NT\(_x\) MXene was obtained by etching Al from its MAX phase V\(_2\)AlN. The MAX phase V\(_2\)AlN was initially obtained by ball milling vanadium and AlN and further cold pressing them to form a pellet. The pellet was subjected to heat treatment thereafter and this step helps in the densification and reducing the number of structural flaws, such as microvoids and cracks. The pressure applied during cold pressing is found to influence the preferred orientation of the grain during its growth. Al from the MAX phase was etched away to synthesize V\(_2\)NT\(_x\) MXene. A schematic showing the flowchart of synthesizing V\(_2\)NT\(_x\) MXene is given in Scheme 1. The as-synthesized V\(_2\)NT\(_x\) MXene was tested as a supercapacitor electrode using an aqueous electrolyte. A specific capacitance of 112.8 F/g was obtained for the electrode at a

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**Scheme 1. Process Flow Indicating the Formation of V\(_2\)NT\(_x\) MXene**

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current density of 1.85 mA/cm² with an energy density of 15.66 Wh/kg and a power density of 3748.4 W/kg.

To further evaluate the potential application of V₂NTₓ as an asymmetric supercapacitor (ASC) device was fabricated using V₂NTₓ MXene as the negative (negative electrode) and Mn₃O₄ nanowalls (NWs) grown on carbon fibers (CFs) as the positive electrode. With the use of aqueous electrolytes in symmetric supercapacitors, the voltage window is limited as the dissociation of water happens at 1.2 V. This drawback could be overcome with the use of organic electrolytes. Although organic electrolytes help in increasing the potential window, they are not safe to use. ASCs help to overcome this issue by operating at higher voltages in an aqueous electrolyte by combining the potential windows of the two different electrode materials in it. Although the operating voltage of an aqueous symmetric supercapacitor is limited to 1.2 V, an ASC with an aqueous electrolyte can operate beyond 2 V.⁵⁷ Generally, carbon is used as the negative electrode with a pseudocapacitive material as a positive electrode in ASCs. The capacitance of the ASC can be increased by replacing the low-capacitive carbon electrode with the MXene electrode.⁵⁸ Among the transition metal oxides, manganese oxide, particularly, Mn₃O₄, has gained limelight because of its high specific capacitance, outstanding structural flexibility, abundant nature, low cost, and environmental benignity.⁵⁹ Wang et al. reported that Mn₃O₄ nanoparticles synthesized through an ultrasound-assisted method exhibited a specific capacitance of 261 F/g at 0.4 A/g.⁶⁰ In the work reported by Qiao et al., porous Mn₃O₄ synthesized via a solvothermal technique exhibited a specific capacitance of 302 F/g at 0.5 A/g with a capacitance retention of 89% even after 5000 cycles.⁶¹ Herein, Mn₃O₄ NWs grown on CF as reported earlier by us² were used as the positive electrode with V₂NTₓ MXene as the negative electrode in the construction of ASCs. The ASC exhibited a cell voltage of 1.8 V in an aqueous KOH electrolyte.

2. RESULTS AND DISCUSSION

To have a systematic analysis, the ball milling time is varied to study the effect of the same in product formation. The mixture of vanadium and AlN was ball-milled for 8, 15, 20, and 26 h and the X-ray diffraction (XRD) pattern obtained for each of them is shown in Figure 1. XRD for the sample obtained after 26 h of milling has well-defined peaks compared to those obtained for 8, 15, and 20 h, and in this regard, the sample ball-milled for 26 h was chosen for further analysis. In the XRD pattern of the sample ball-milled for 26 h, peaks of vanadium (JCPDS card no: 00-022-1058) and AlN (JCPDS card no: 00-025-1133) could be identified.

Figure 2 shows the comparison of XRD obtained for MAX phase V₂AlN, the V₂NTₓ obtained after treating with an acid–salt mixture and V₂NTₓ obtained after delamination by treating with DMSO. From Figure 2, it can be observed that the XRD pattern obtained for MAX phase V₂AlN is different from that for the carbide MAX phases as there are no peaks (002) in the lower angles. After the MAX phase V₂AlN is subjected to treatment with the LiF–HCl mixture, it can be seen that the number of peaks increased and this can be attributed to the fluoride salt impurities and Al₂O₃ formed during the etching process. Peaks corresponding to Al₂O₃ (JCPDS card no: 00-042-1468) are present in the material, although less intense, because aluminum tends to get easily oxidized leading to the formation of Al₂O₃. The XRD pattern of V₂NTₓ MXene subjected to delamination indicates that most of the MAX phase peaks are greatly diminished in intensity and the peaks are slightly shifted to the left.

The morphology of the samples was examined by field emission scanning electron microscopy (FESEM) imaging. The FESEM image of the MAX phase V₂AlN is shown in Figure 3a. Although the layered structure is visible to some extent, they are piled up into blocks of uneven sizes, surface edges, and distribution. The elemental analysis of the sample showed the presence of vanadium, aluminum, and nitrogen in a weight ratio of 2:1:1, respectively (Figure 3b). The elemental mapping in detail is given in Figure S1, which shows the uniform distribution of vanadium, aluminum, and nitrogen. The FESEM image of LiF–HCl-treated V₂AlN is given in Figure 3c. The LiF–HCl solution is capable of etching aluminum from V₂AlN to form a multilayered V₂NTₓ MXene. The V₂NTₓ MXene here do not possess a perfect lamellar structure as that of carbide MXenes reported but it depicts the formation of 2D layers which are still stacked with uneven edges. The energy-dispersive X-ray spectroscopy (EDAX) spectrum of the acid–salt-treated V₂AlN is shown in Figure 3d. The spectrum shows the presence of surface-terminating groups such as −F and −O as a resultant of acid–salt treatment. The aluminum peak indicates that it is not completely etched after the acid–salt treatment. However,
the intensity of the aluminum peak is reduced comparatively. The presence of the carbon peak is due to the use of conductive carbon tape during the analysis. The elemental mapping in detail is given in Figure S2.

Figure 4a–b shows the typical 2D sheet structures obtained after V$_2$NT$_x$ was treated with DMSO.

Perfect lamellar structures are now visible but with no complete delamination and this could be due to the presence of tiny amounts of Al in the material. The EDAX spectrum of the layered V$_2$NT$_x$ MXene is given in the inset of Figure 4b. The intensity of the aluminum peak in the EDAX spectrum is further reduced indicating its loss during the exfoliation process. To evaluate the amount of aluminum left on the MXene sheets, inductively coupled plasma–optical emission spectrometry (ICP–OES) analysis was carried out. The percentage amount of aluminum present in the global sample was 9.26%, which was about six times lesser than that of vanadium. This proves the successful formation of V$_2$N from V$_2$AlN. The color-coded elemental mapping indicates the uniform distribution of the elements (Figure S3). In order to view the sheets clearly and to further confirm the morphological characteristics, transmission electron microscopy (TEM) analysis was carried out and the images are shown in Figure 4c–d.

The composition and the electronic states of the delaminated V$_2$NT$_x$ MXene were investigated using X-ray
photoelectron spectroscopy (XPS). CasaXPS software with a shirley background type was used in the XPS peak fitting. The core-level XPS V2p and N1s spectra are depicted in Figure 5.

The V2p regions overlap with the O1s region because of the presence of oxygen on the sample surface and hence they are fitted together. The characteristic peaks at 515.4 and 522.8 eV are ascribed to V2p3/2 and V2p1/2, respectively. The high intensity peak of O1s at 529 eV indicates the high concentration of oxygen onto the surface of vanadium.43 With the absence of vanadium oxide peaks in the XRD pattern, it can further be confirmed that functionalization of oxygen has occurred only on the surface and not in the bulk sheets.44

The V2NT_x MXene supercapacitor electrodes were examined by slurry-coating V2NT_x MXene onto CF cloth. The CF cloth is electrically conducting and serves as the current collector for the supercapacitor electrode. The supercapacitive performances of the V2NT_x MXene electrode are examined using electrochemical impedance spectroscopy (EIS), cyclic voltammetry (CV), and galvanostatic charge–discharge (GCD) measurements. The Nyquist plot of the V2NT_x MXene electrode is depicted in Figure 6a and the inset shows the Nyquist plot at the magnified high-frequency region. From the Nyquist plot, it can be observed that the V2NT_x MXene electrode exhibits an electrochemical series resistance (ESR) of 0.4 Ω. The CV curves for the V2NT_x MXene electrode obtained at different scan rates within a potential window of −1.0 to 0 V are depicted in Figure 6b. The CV curves display nonrectangular profiles with redox peaks which is an evidence of pseudocapacitive charge storage exhibited by the V2NT_x MXene because of its surface functional groups. The V2NT_x MXene electrode undergoes faradaic reactions with the

Figure 5. Deconvoluted high-resolution XPS spectra of V2p and N1s.

Figure 6. (a) Nyquist plot obtained in a frequency range of 10^6 to 0.02 Hz (inset shows the Nyquist plot in the high-frequency region); (b) CV curves obtained at different scan rates; (c) GCD curves obtained at different current densities of the V2NT_x MXene electrode; (d) plot of specific capacitance and (e) energy density at different current densities of the V2NT_x MXene electrode; and (f) plot of capacitance retention at different cycle numbers of the V2NT_x MXene electrode.
electrolyte thereby enhanced charge storage can be obtained. This is due to the presence of functional groups (−F and −O) on the surface of V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene. Figure 6c shows the GCD curves of the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode obtained at different current densities at a potential window of −1.0 to 0 V. The faradaic charge storage possessed by the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode is evident from the GCD curves too as it is different from the linear GCD profiles for a typical electrochemical double layer capacitor. The specific capacitance of the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode is calculated from these GCD curves and plotted in Figure 6d. A specific capacitance of 112.8 F/g is obtained for the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode at a current density of 1.85 mA/cm<sup>2</sup>. The capacitance is found to decrease gradually at higher current densities, which is typical for the pseudocapacitors. The energy density at different current densities of the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode is plotted in Figure 6e. An energy density of 15.6 W h/kg was obtained at a current density of 1.85 mA/cm<sup>2</sup> with the corresponding power density of 3748.4 W/kg. However, the electrode displayed a high power density of 14,987.4 W/kg at a current density of 7.5 mA/cm<sup>2</sup>. In order to examine the electrochemical cycling stability of the electrode, a GCD study was performed for continuous 10,000 cycles at a constant current density of 5 mA/cm<sup>2</sup>. The variation in the specific capacitance retention at different cycle numbers is plotted in Figure 6f. A capacitance retention of 96% is observed even after 10,000 cycles, which shows the excellent electrochemical cycling stability of the V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene electrode. It is well-known that nitride MXenes exhibit remarkable electrochemical stabilities and thus are potential candidates for application in supercapacitors with long cycle life.

An ASC device was fabricated to expand the operating voltage window while using an aqueous electrolyte. It is well-known that a cell voltage >1.2 V cannot be obtained in a symmetric configuration because of the water dissociation. However, the cell voltage can be increased beyond this point by fabricating an ASC device, which utilizes dissimilar electrodes in an aqueous electrolyte. This strategy of fabricating ASCs has advantages of manufacturing supercapacitors with high working voltages in a safe aqueous electrolyte; hence, the flammable organic electrolytes can thus be avoided. We have fabricated an ASC device using V<sub>2</sub>N<sub>T</sub><sub>x</sub> MXene as the negative electrode and Mn<sub>3</sub>O<sub>4</sub> NWs grown on CF as the positive electrode and tested in a two-electrode cell configuration using a 3.5 M KOH aqueous electrolyte. The use of Mn<sub>3</sub>O<sub>4</sub> NWs grown on CF as an electrode in a symmetric supercapacitor is reported previously. The scanning electron microscopy (SEM) image of Mn<sub>3</sub>O<sub>4</sub> NWs grown on CF is shown in Figure 7a from which it can be observed that the Mn<sub>3</sub>O<sub>4</sub> NWs are grown on CF uniformly with a porous...
architecture. In order to understand the electrochemical performance of the Mn₃O₄ NWs, the CV study is conducted within a potential window of 0–0.8 V in a 3.5 M KOH aqueous electrolyte. The CV curves obtained at different scan rates of the Mn₃O₄ NW electrode are shown in Figure 7b. The curves display typical faradaic charge storage possessed by the Mn₃O₄ NWs with distinguished reduction/oxidation peaks. The faradaic charge storage is helpful in achieving high specific capacitance and energy densities. The supercapacitive performances of the Mn₃O₄ NWs/V₂NTₓ MXene ASC device are examined by EIS, CV, and GCD measurements. The Mn₃O₄ NWs/V₂NTₓ MXene ASC device was able to operate within a potential window of −1.0 to 0.8 V, thereby obtaining a cell voltage of 1.8 V. The Nyquist plot of the ASC device is depicted in Figure 7c and the inset shows the Nyquist plot in the magnified high-frequency region. From the Nyquist plot, it can be observed that the ASC device exhibits an ESR of 1.5 Ω.

4. EXPERIMENTAL SECTION

4.1. Experimental Reagents. Elemental powders of vanadium (325 mesh, 99.5%) and aluminum nitride (AlN; 10 μm, ≥98%), lithium fluoride (LiF, trace metals basis ≥99.9%), isopropyl alcohol (IPA), and DMSO (≥ 99.9%) were purchased from Sigma-Aldrich, USA. Super P carbon black and poly(vinylidene fluoride) (PVDF) were supplied by Alfa Aesar, USA.

4.2. Synthesis of V₂AlN. The MAX phase V₂AlN was prepared initially by ball milling the elemental powders of vanadium and AlN at a molar ratio of 2:1 with zirconia balls with a powder-to-ball ratio of 1:20 for 26 h at 250 rpm. The ball-milled precursor was pressed hydraulically to obtain a pellet by applying a pressure of 4 tons. The resultant pellet was maintained at a temperature of 1100 °C for 6 h in a tubular furnace with a ramping rate of 5°/min. The heating was carried out under an argon atmosphere. A solid brick-like pellet was obtained, which was ground thoroughly to form fine powders.

4.3. Synthesis of V₂NTₓ. An acid–salt etching of Al with LiF–HCl was carried out as reported previously. In a typical procedure, 0.32 g of LiF was dissolved in 6 M HCl by magnetic stirring for 20 min. To this acid–salt mixture, 0.5 g of the V₂AlN powder was added slowly. This reaction mixture was kept undisturbed for 3 h and during this time, a vigorous reaction between the acid–salt mixture and V₂AlN was observed. Later, the reaction mixture was sonicated for an hour at 40°C and centrifuged several times with water until the supernatant maintained a pH of 6. Finally, the precipitate settled at the bottom of the centrifuge tube was washed with IPA at 4000 rpm and vacuum-filtered to obtain V₂NTₓ MXene. The obtained MXene was subjected to delamination by hand-shaking with DMSO for 5 min and then sonicating for 1 h. The solution was then kept undisturbed overnight, centrifuged, and filtered.

4.4. Synthesis of Mn₃O₄ NWs on CF. The electrochemical deposition method was used in the synthesis of two-dimensional Mn₃O₄ NWs on the CF substrate as per the literature. In a typical procedure, an electrochemical deposition bath consisting of 0.16 M Na₂SO₄ and 0.16 M manganese acetate tetrahydrate was used in a three-electrode cell in which a surface-pretreated CF substrate was used as the working electrode, Ag/AgCl was used as the reference electrode, and platinum foil was used as the counter electrode. Mn₃O₄ NWs were deposited at a constant potential of −1.8 V for a period of 20 min through the chronoamperometry technique. The as-synthesized Mn₃O₄ NWs on CF were further washed and dried. Mn₃O₄ NWs grown on CFs were used as the positive electrode for fabricating an ASC in this study.

4.5. Preparation of the V₂NTₓ MXene Electrode and Fabrication of V₂NTₓ //Mn₃O₄ NW ASC. The V₂NTₓ MXene electrode was prepared using the as-synthesized V₂NTₓ MXene powder as the electrode-active material. Initially, the V₂NTₓ MXene powder, Super P carbon black, and PVDF were taken at a weight ratio of 85, 10, and 5 wt %, respectively, and a slurry was prepared by mixing them together in NMP. The slurry was then coated on to the CF substrate with a geometric area of 2 × 2 cm² and dried in an oven thereafter. Here, the CF was used as the current collector. An ASC device was fabricated using V₂NTₓ MXene as the negative electrode and Mn₃O₄ NWs as the positive electrode.

4.6. Characterization. XRD analysis was carried out to examine the crystal structure of V₂AlN and V₂NTₓ MXene using a Bruker D8 ADVANCE X-ray Diffractometer with Cu Kα radiation. The morphology of the V₂NTₓ MXene was examined using a field emission scanning electron microscope imaging (Zeiss) operated at 10 kV equipped with EDAX. The
The CV study of the Mn$_3$O$_4$/CF electrode was performed at various current densities such as 1.85, 2.50, 3.75, 5.00, 6.25, and 7.50 mA/cm$^2$. The specific capacitance was calculated using eq 1.

$$C_p = 2 \times \frac{I \times \Delta t}{V \times m}$$

where $C_p$ is the specific capacitance (in F/g), $I$ is the current (in A), $\Delta t$ is the discharging time (in s), $m$ is the active mass of the electrode-active material (in g), and $V$ is the operating voltage window (in V) of the electrode/ASC device. The energy and power densities were calculated using the following eqs 2 and 3, respectively.

$$E = \frac{1}{2}C_p(\Delta V)^2$$

$$P = \frac{E}{\Delta t}$$

where $E$ is the energy density and $P$ is the power density and all other variables are the same as discussed previously.

**4.7. Electrochemical Measurement.** The electrochemical behavior of the V$_2$NT$_x$ MXene electrode was examined in a three-electrode cell configuration using a 3.5 M KOH aqueous electrolyte. The V$_2$NT$_x$ MXene, Ag/AgCl, and Pt foil were used as the working, reference, and counter electrodes, respectively. The EIS analysis of the V$_2$NT$_x$ MXene electrode was carried out in a frequency range of 10$^6$ to 0.02 Hz. The CV study of the V$_2$NT$_x$ MXene electrode was performed within a potential window of −1.0 to 0 V versus Ag/AgCl at different scan rates such as 100, 80, 60, 40, 20, and 10 mV/s. The GCD measurements of the V$_2$NT$_x$ MXene electrode were carried out in a potential window of −1.0 to 0 V at different current densities such as 1.85, 2.50, 3.75, 5.00, 6.25, and 7.50 mA/cm$^2$. The CV study of the Mn$_3$O$_4$/CF electrode was performed within a potential window of 0−0.8 V. The Mn$_3$O$_4$ NWs/V$_2$NT$_x$ MXene ASC device was tested in a two-electrode configuration using a 3.5 M KOH aqueous electrolyte. The EIS study of the ASC device was carried out in a frequency range of 10$^6$ to 0.02 Hz, and the CV study was performed within a potential window of −1.0 to 0.8 V at different scan rates such as 100, 80, 60, 40, 20, and 10 mV/s. The GCD measurement of the ASC device was carried out in a potential window of −1.0−0.8 V at different current densities such as 1.85, 2.50, 3.75, 5.00, 6.25, and 7.50 mA/cm$^2$. From the GCD curves, the specific capacitance, energy density, and power density were calculated. The specific capacitance was calculated using eq 1.

**ASSOCIATED CONTENT**

**Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsomega.0c01215.

FESEM images and the elemental mapping of V$_2$AlN; LiF−HCl-treated V$_2$AlN; and delaminated V$_2$AlN (PDF)

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**Notes**

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17990

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