Magne

MAGNETOSTRUCTURAL TRANSITION, METAMAGNETISM, AND MAGNETIC PHASE COEXISTENCE IN CO_{10}Ge_{3}O_{16}

Phillip T. Bartok and Ram Seshadri
Materials Department and Materials Research Laboratory,
University of California, Santa Barbara, CA, 93106, USA

Matthew R. Suchomel
X-Ray Science Division, Argonne National Laboratory, Argonne, IL 60439, USA
(Dated: May 11, 2014)

CO_{10}Ge_{3}O_{16} crystallizes in an intergrowth structure featuring alternating layers of spinel and rock salt. Variable-temperature powder synchrotron X-ray and neutron diffraction, magnetometry, and heat capacity experiments reveal a magnetostructural transition at \( T_N = 203 \) K. This rhombohedral-to-monoclinic transition involves a slight elongation of the COGe octahedra along the apical axis. Below \( T_N \), the application of a large magnetic field causes a reorientation of the Co^{2+} Ising spins. This metamagnetic transition is first-order as evidenced by a latent heat observed in temperature-dependent measurements. This transition is initially seen at \( T = 180 \) K as a broad upturn in the \( M-H \) near \( H_C = 3.9 \) T. The upturn sharpens into a kink at \( T = 120 \) K and a “butterfly” shape emerges, with the transition causing hysteresis at high fields while linear and reversible behavior persists at low fields. \( H_C \) decreases as temperature is lowered and the loops at positive and negative fields merge beneath \( T = 20 \) K. The antiferromagnetism is described by \( k_M = (000) \) and below \( T = 20 \) K a small uncompensated component with \( k_M = (000) \) spontaneously emerges. Despite the Curie-Weiss analysis and ionic radius indicating the Co^{2+} is in its high-spin state, the low-temperature \( M-H \) trends toward saturation at \( M_S = 1.0 \) \( \mu_B/Co \). We conclude that the field-induced state is a ferrimagnet, rather than a \( S = 1/2 \) ferromagnet. The unusual \( H-T \) phase diagram is discussed with reference to other metamagnets and Co(II) systems.

PACS numbers: 75.30.Kz 75.50.Ee 75.60.Nt 75.50.Gg

I. INTRODUCTION

Compounds featuring Co^{2+} often display interesting magnetic field-induced phase transitions.\cite{1} The driving force behind these transitions is the Ising-like single-ion anisotropy of octahedral Co^{2+} that results from spin-orbit-lattice coupling. Co^{2+} is \( d^7 \) and in an octahedral coordination environment it typically adopts a high-spin \( 4\Gamma_{9/2} \) electronic ground state (\( S = 3/2 \) and \( L = 3 \)); with spin-orbit coupling it is often described as a Kramer’s doublet with \( J_{\text{eff}} = 1/2 \). The complex magnetism that results from this electronic state is seen in many types of Co(II)-containing antiferromagnets including the prototypical CoCl\(_2\)·2H\(_2\)O\cite{2} β-Co(OH)\(_2\)\cite{3} hybrid metal-organic frameworks\cite{4} and oxyselenides.\cite{5} Co(II) is also a popular species to study in the field of molecular magnets where it is of interest for spin-crossover or spin-state transitions.\cite{6}

The majority of the aforementioned anisotropic metamagnets exhibit spin reorientations and their phase diagrams are well described by an Ising model with magnetic exchange that is antiferromagnetic for nearest neighbors and ferromagnetic for next-nearest neighbors.\cite{7,8} These anisotropic systems are distinct from those with Heisenberg spins which undergo spin-flop transitions where spin canting is important. Field-induced transitions also occur in systems with ferrimagnetic or noncollinear ground states such as CoCr\(_2\)O\(_4\).\cite{9} These ground states are not as well investigated as Néel antiferromagnetism, but are more technologically interesting. Some of these systems have complex \( H-T \) phase diagrams that have been explored by experimentalists and are beginning to be explained by theorists. In this manuscript, we investigate the complex behavior of the relatively unstudied \( \text{Co}_{10}\text{Ge}_{3}\text{O}_{16} \).

\( \text{Co}_{10}\text{Ge}_{3}\text{O}_{16} \) is an octahedral Co^{2+} compound that is related to the spinel GeCo\(_2\)O\(_4\) in both chemical composition and crystal structure. It was first prepared by Barbier who solved its structure from single-crystal X-ray diffraction.\cite{10} Its layered crystal structure is an intergrowth composed of alternating spinel units and edge-sharing octahedra layers (Figure 1). The O\(^{2-} \) anions are close-packed, with a nearly ideal c/a ratio, and the Ge\(^{4+} \) sit in both tetrahedral and octahedral sites. Barbier’s study found that bond valence sums indicate the expected ionic states and that the thermal parameters are almost isotropic. This structure type is related to the cation-deficient mineral aerugite, \( \text{Ni}_{8.5}\text{As}_{0}\text{O}_{16} \).\cite{11,12} Udod and co-workers concluded that \( \text{Co}_{10}\text{Ge}_{3}\text{O}_{16} \) is a ferrimagnet based on magnetometry measurements,\cite{13} however our study indicates that the exceedingly complex magnetism is not captured by the term “ferrimagnetism”.

Here we present physical property measurements in-
II. METHODS

Polycrystalline pellets of Co$_{10}$Ge$_3$O$_{16}$ were prepared by solid-state reaction of powder reagents at high temperature. Stoichiometric amounts of GeO$_2$ and Co$_3$O$_4$ were ground with an agate mortar and pestle, pressed at 100 MPa, and fired in air at 1000°C for 48 h with an intermediate grinding. The pellets were placed on beds of powders of the same composition to avoid contamination by the ZrO$_2$ crucible. Powder synchrotron X-ray diffraction was performed at the 11-BM beamline ($\lambda = 0.413104$ Å) of the Advanced Photon Source, Argonne National Laboratory. Powder neutron diffraction was conducted at the HIPD beamline of the Lujan Neutron Scattering Center, Los Alamos National Laboratory. Rietveld($^{15}$) refinements were performed using GSAS/EXPGUI($^{13}$) DICVOL, as implemented in FullProf, was used to index the low-temperature unit cell($^{15}$) ISODISTORT was used to explore the possible crystal distortion modes and to transform the unit cell atom positions to lower symmetry.$^{[22]}$ Crystal structures were visualized using VESTA.$^{[18]}$ Magnetic properties were measured using a Quantum Design MPMS 5XL SQUID magnetometer and a PPMS DynaCool VSM. Heat capacity was measured using a thermal relaxation method as implemented in a Quantum Design PPMS. Electrical resistivity was measured by a PPMS DynaCool ETO to be 5×10$^6$ Ω·cm at room-temperature and increased exponentially as temperature decreased, consistent with the expected insulating behavior of a brown-colored material.

III. RESULTS AND DISCUSSION

A. Crystal structure

Variable-temperature powder synchrotron X-ray diffraction was used to investigate the crystal structure of Co$_{10}$Ge$_3$O$_{16}$. The known crystal structure was confirmed by Rietveld refinement of the room-temperature diffraction pattern (not shown). The extracted room-temperature $R\bar{3}m$ structural parameters are $a = 5.957(2)$ and $c = 28.92(2)$, which agree well with the single-crystal study by Barbier.$^{[16]}$ The presence of a small 1.0 mol% impurity of the GeCo$_2$O$_4$ phase was determined by our analysis. An unidentified and even smaller impurity phase was observed in the synchrotron X-ray data with peaks at $Q = 1.34(4), 1.42(8), 1.45(7), 1.46(7), 2.38(8), \text{ and } 3.10(9)$ Å$^{-1}$. We rule out any significant impurity effect on physical properties based on their small phase fractions as well as experiments on a second sample that exhibits the same behaviors yet does not
contain any observable impurities. A structural phase transition was observed in the temperature evolution of the powder synchrotron X-ray diffraction. The onset of the transition is indicated by the splitting of the \( R\bar{3}m \) (208) diffraction peak as temperature is lowered (Figure 2). The diffraction data were fit by Rietveld refinement using the known \( R\bar{3}m \) model above the transition and a new \( C2/m \) model below. Unit cell parameters for the initial \( C2/m \) model were determined by indexing the diffraction pattern. The \( C2/m \) atom positions were derived from the \( R\bar{3}m \) model using group-subgroup theory. A refinement using the \( C2/m \) model at \( T = 100 \) K gives a good qualitative fit to the data, matching the observed peaks in both position and intensity (Figure 3). The small residual and R\textsubscript{Bragg} support the validity of both the new low-temperature \( C2/m \) model and the known high-temperature \( R\bar{3}m \) model. The extracted structural parameters for \( C2/m \) at \( T = 100 \) K are listed in Table I. The structural transition was determined to occur between \( T = 200 \) and \( 210 \) K by comparing fits of the diffraction data using the high- and low-temperature structural models. This temperature is consistent with the \( T_N = 203 \) K from the magnetometry experiments discussed later. The unit cell parameters decrease smoothly with temperature, as expected for a typical positive coefficient of thermal expansion (Figure 4). A change of slope, but not a discontinuity, is observed at \( T_N \) in the plot of cell volume as a function of temperature; this is typical of a second-order phase transition.

B. Magnetism

The magnetic behavior of \( \text{Co}_{10}\text{Ge}_3\text{O}_{16} \) involves phase transitions with both temperature and magnetic field. Magnetometry data, including both temperature (\( \chi-T \), Figure 5) and field (\( M-H \), Figure 6) dependences, were collected from \( T = 1.8 \) K to 380 K under applied magnetic fields up to \( H = 9 \) T. A standard zero-field cooled (ZFC) and field-cooled (FC) protocol was used for the temperature-dependent measurements. A Curie-Weiss fit of the magnetic susceptibility collected under \( H = 1000 \) Oe in the paramagnetic regime from \( T = 1.8 \) K to 380 K gives \( \mu_{\text{eff}} = 4.26 \) \( \mu_B/\text{Co} \), assuming \( g = 2 \), and \( \Theta_{\text{CW}} = +39.3 \) K. The magnitude of \( \mu_{\text{eff}} \) is consistent with
TABLE I: Structural parameters of \( \text{Co}_{10}\text{Ge}_5\text{O}_{16} \) at \( T = 100 \text{K} \), as determined by Rietveld refinement of powder synchrotron X-ray diffraction data. Space group: \( \text{C}2/m \), \( a = 10.3021(0) \text{ Å} \), \( b = 5.9537(4) \text{ Å} \), \( c = 10.2158(1) \text{ Å} \), and \( \beta = 109.5(4) ^\circ \). Figures of merit: \( \chi^2 = 1.93 \), \( R_{wp} = 6.40 \% \), \( R_p = 5.07 \% \).

| Site | \( x \) | \( y \) | \( z \) | \( U_{iso} (\text{Å}^2) \) |
|------|---------|---------|---------|-----------------|
| Ge1  | 0       | 0       | 0       | 0.001(1)        |
| Ge2  | 0.1892(9) | 0       | 0.5679(4) | 0.001(5)        |
| Co1  | 0       | 0.5     | 0       | 0.004(0)        |
| Co2  | 0.25    | 0.25    | 0       | 0.001(7)        |
| Co3  | 0       | 0.5     | 0       | 0.002(5)        |
| Co4  | 0.0039(3) | 0.2451(0) | 0.2562(9) | 0.001(1)        |
| Co5  | 0.2483(3) | 0       | 0.2558(3) | 0.001(1)        |
| O1   | 0.1266(8) | 0       | 0.3821(8) | 0.003(7)        |
| O2   | 0.3766(2) | 0       | 0.1283(1) | 0.004(7)        |
| O3   | 0.6136(7) | 0.2761(1) | 0.1146(2) | 0.003(2)        |
| O4   | 0.8863(1) | 0       | 0.1118(8) | 0.002(0)        |
| O5   | 0.3728(3) | 0.2548(7) | 0.3741(0) | 0.003(9)        |
| O6   | 0.6293(9) | 0       | 0.3775(9) | 0.005(0)        |

FIG. 5: (Color online) Magnetic susceptibility as a function of temperature for \( \text{Co}_{10}\text{Ge}_5\text{O}_{16} \). The data were collected under \( H = 0.1, 1, \) and \( 5 \text{T} \) using a ZFC-FC procedure.

high-spin \( \text{Co}^{2+} \), lying between its \( S \) and \( L + S \) values of 3.87 and 5.20. The positive \( \Theta_{\text{CW}} \) implies the presence of ferromagnetic interactions. However, as is known for \( \text{GeCo}_2\text{O}_4 \), the Curie-Weiss law does not strictly apply for octahedral \( \text{Co}^{2+} \) because of low-lying excited states that affect the susceptibility. Consequently, the sign of \( \Theta_{\text{CW}} \) could be misleading. In low fields of \( H = 100 \) (not shown) and 1000 Oe, a cusp in the \( \chi-T \) occurs at \( T_N = 203 \text{K} \), suggesting that long-range antiferromagnetic order is established. Below this \( T_N \), a small broad hump, centered at \( T = 170 \text{K} \), and a slight ZFC-FC splitting are observed. \( \chi \) reaches a local minimum at \( T = 100 \text{K} \) and then increases as temperature is further lowered, with a sharp upturn at \( T = 20 \text{K} \). Considering only the FC trace, this increase in \( \chi \) might be interpreted as a Curie tail from a paramagnetic impurity. However, we also observe that the ZFC-FC splitting increases significantly below \( T = 20 \text{K} \) and that the ZFC exhibits a peak at \( T = 9 \text{K} \). These features in the ZFC-FC indicate the onset of another phase transition involving uncompensated magnetism. As seen in the \( H = 1 \) and \( 5 \text{T} \) susceptibility curves, this low-temperature transition is strongly affected by the application of a large magnetic field, the details of which are discussed in the next paragraph.Isothermal \( M-H \) sweeps are shown in Figure 5 for different temperatures down to \( T = 1.8 \text{K} \). The curves are separated into three groups by temperature region and plotted in sequence to simulate the evolution with temperature. The \( M-H \) is linear above the transition at \( T = 220 \text{K} \), consistent with paramagnetism at low \( B/T \). As temperature is decreased below \( T_N = 203 \text{K} \), the \( M-H \) behavior remains linear with the slope following that expected from the \( \chi-T \). However, a broad upturn in the \( M-H \) at high field begins to evolve as \( T \) passes beneath \( 180 \text{K} \). By \( T = 120 \text{K} \), this upturn develops into a sharp kink at a critical field \( H_C = 3.9 \text{T} \). This phase transition involves hysteresis at high fields while the dependence remains linear and reversible at low fields. A “butterfly” shape is thus formed, with distinct loops existing at both high positive and negative fields. \( H_C \) decreases as temperature is further lowered and the hysteresis loops eventually merge starting at \( T = 20 \text{K} \). The \( M-H \) sweep at \( 1.8 \text{K} \) exhibits a single hysteresis loop with a sizable coercivity of nearly 1.5T. Interestingly, the virgin magnetizing curve taken after ZFC lies outside the hysteresis loop. This unusual \( M-H \) feature has previously been associated with irreversible domain wall motion in spinel oxides. Looking back at the \( \chi-T \), we see that the measurement under \( H = 5 \text{T} \) exhibits a change in slope at \( T_N \), as opposed to a cusp, and that \( \chi \) has a larger magnitude than in small fields, consistent with \( H_C < 5 \text{T} \).

The application of a large magnetic field generates a first-order phase transition in \( \text{Co}_{10}\text{Ge}_5\text{O}_{16} \) at temperatures below \( 120 \text{K} \). This corresponds to the same temperature at which the field-induced transition in the \( M-H \) becomes sharp and hysteric. The first-order nature of the transition was first suspected because of an unusual feature seen in the high-field susceptibility data where the ZFC trace is larger in magnitude than the FC (Figure 5). For measurement under \( H = 1 \) this ZFC > FC for a region \( 7 \text{K} < T < 30 \text{K} \), while it is \( 14 \text{K} < T < 120 \text{K} \) for \( H = 5 \text{T} \). These data were collected using a normal ZFC-FC procedure where the ZFC is collected on warming and the FC is collected on cooling, however following this unusual ZFC > FC observation, we measured a second FC curve on heating to investigate any history effect. A thermal hysteresis, indicating a latent heat, is observed between the FC data collected on heating and cooling, with the curves offset in the \( H = 1 \text{T} \) measurement by 11 K at the largest point (Figure 7). This latent heat is only observed for measurements with \( H > 0.3 \text{T} \). For \( H = 1 \text{T} \), the bifurcation point between...
FIG. 6: (Color online) Magnetization versus field sweeps for \(\text{Co}_{10}\text{Ge}_3\text{O}_{16}\) at different temperatures. The data for each temperature are offset by 5 T increments along the x-axis to facilitate visualization. The virgin curve for the \(T = 1.8\) K trace is dashed to distinguish it from the loop. Dashed black lines are provided to establish the graph origin for each data set.

FIG. 7: (Color online) A field-induced first-order phase transition at low temperature in \(\text{Co}_{10}\text{Ge}_3\text{O}_{16}\). The zero-field cooled (ZFC) trace is greater than the field-cooled (FC) for \(7 < T < 30\) K. This is a result of thermal hysteresis between the FC data collected on heating and cooling, with the curves offset by 11 K at the largest point.

Heat capacity measurements were used as a complementary method to investigate the phase transitions (Figure 8). An anomaly is observed at \(T_N = 203\) K under zero applied field, consistent with the magnetostructural transition seen in the diffraction and magnetometry data. The field dependence of the high-temperature measurements reveals that the transition temperature is decreased by a few Kelvin upon the application of \(H = 7\) T, consistent with antiferromagnetic theory. No obvious features are seen down to \(T = 2\) K in the zero-field trace, nor in the measurements under \(H = 1\) or \(7\) T, despite the presence of the field-induced first-order transition revealed by magnetometry. However, the \(C_p/T\) shows an upturn below 15 K that is suppressed with field. It is difficult to assign this second anomaly to a specific magnetic, structural, or electronic origin. Its weak nature suggests that the low-temperature magnetic transition is a spin reorientation and thus does not involve a change in entropy. The entropy loss could instead be a result of a structural transition; a subtle change in octahedral shape, which is already distorted beneath \(T_N\), as opposed to an order-disorder transition for example, would gen-
FIG. 9: (Color online) Powder neutron diffraction for Co$_{10}$Ge$_3$O$_{16}$ at different temperatures. Magnetic Bragg reflections of $k_M = (00\frac{1}{2})$ emerge beneath $T_N = 203$ K. The peak positions of the nuclear and $k_M = (00\frac{1}{2})$ magnetic phases are shown in the panel above the patterns, colored red and black respectively. The $T = 5$ and 100 K patterns are quite similar, however a close look reveals a new magnetic peak at $Q = 0.65$ Å$^{-1}$, the (001) nuclear position, as well as additional magnetic intensity at other nuclear reflections, which is consistent with the onset of an additional $k_M = (000)$. This development occurs for $T < 20$ K and is associated with the uncompensated magnetism that is observed concomitantly. Each panel displays data from a separate detector bank with different resolution.

...erate only a small response. It is also possible that this measurement is probing the Kramers’ doublet electronic ground state of octahedral Co(II). Finally, the first-order nature is problematic for the thermal relaxation mechanism of our heat capacity measurement. We are thus planning to investigate the heat capacity with a technique that directly measures heat flux.

Powder neutron diffraction (PND) was conducted to further elucidate the magnetism of Co$_{10}$Ge$_3$O$_{16}$ (Figure [9]). We do not report the magnetic structure here, however we discuss the implications of some important observations. Magnetic Bragg reflections materialize beneath $T_N = 203$ K and confirm the onset of antiferromagnetism. The structural component of the phase transition known from the synchrotron X-ray experiments is just barely evident as a subtle peak broadening in the neutron data. The peak positions index to a magnetic $k$-vector $k_M = (00\frac{1}{2})$, indicating that the magnetic unit cell is commensurate with the low-temperature monoclinic nuclear structure but that it is doubled along the $c$-axis. The intensities and positions of the peaks evolve smoothly with temperature down to $T = 20$ K at which point a small new peak emerges at $Q = 0.65$ Å$^{-1}$, along with some subtle modulations of intensity in other reflections. We attribute this new reflection to a magnetic origin because although its $Q$ value corresponds to the (001) plane of the nuclear unit cell, and thus it could be a nuclear peak, no intensity is observed above background at this location for $T > 20$ K. A careful comparison of the $T = 5$ and 100 K patterns reveals that magnetic intensity is also added to other solely nuclear reflections such as $Q = 1.24, 1.96, 3.41$ Å$^{-1}$. Additional support for a magnetic origin is seen in the low-field magnetometry data, where a FC increase, ZFC-FC separation, and ZFC peak are observed with the same onset temperature as this new magnetic peak. The occurrence of this magnetic Bragg reflection at the nuclear (001) suggests that it belongs to $k_M = (000)$. Given the uncompensated ferri- or ferromagnetic nature of this $k_M = (000)$ seen in the low-field magnetometry data, we suggest that it is connected to the transition occurring at large magnetic field. We thus hypothesize that the application of a magnetic field will enhance $k_M = (000)$ at the expense of $k_M = (00\frac{1}{2})$.

The spontaneous, and likely field-induced, $k_M = (000)$ uncompensated magnetism of Co$_{10}$Ge$_3$O$_{16}$ appears to be ferrimagnetic in nature. We observe that $M = 0.82 \mu_B$/Co at $T = 1.8$ K and $H = 9$ T, and that the $M$-$H$ is trending toward saturation at an intermediate $M_S = 1.0 \mu_B$/Co, one third of the expected value for $S = 3/2$ Co(II). This value of $M_S$ suggests a ferrimagnetic state where partial compensation of the moment arises from the magnetic structure. An alternative explanation to explain the reduced $M_S$, however, is that Co$_{10}$Ge$_3$O$_{16}$ adopts a ferromagnetic low-spin $S = 1/2$ state. While low-spin Co(II) is found in CoS$_2$ and CoSe$_2$ where the crystal field is larger[21], a spin transition is unheard of in extended, non-molecular Co(II) systems and would involve significant changes in crystal structure. Additionally, given the large magnetic field needed to drive a spin-state transition in the related Co(III) compound LaCoO$_2$[22] a similar transition seems unlikely to occur under our experimental conditions. The spontaneous emergence of this ferrimagnetism and its coexistence with the antiferromagnetism is unusual for Co(II)-containing oxides and even for inorganic compounds in general.

The structural and magnetic $H$-$T$ phase diagram of Co$_{10}$Ge$_3$O$_{16}$, as determined by analysis of variable-temperature powder synchrotron X-ray and neutron diffraction, magnetometry, and heat capacity experiments, is presented in Figure[10]. The free energy land-
the $a$ or $c$ axes. However, single-crystal measurements on the related GeCo$_2$O$_4$ show that it does not exhibit significantly anisotropic magnetism despite the single-ion anisotropy of Co(II). It is instead likely that the magnetic anisotropy is intimately related to the magnetic structure, and thus the large phase space of Co$_{10}$Ge$_3$O$_{16}$ and GeCo$_2$O$_4$ may be a consequence of their high density of magnetic ions and the presence of many exchange pathways.

IV. CONCLUSIONS

The crystal structure of Co$_{10}$Ge$_3$O$_{16}$ is an intergrowth with alternating layers of spinel and rock salt. A magnetostuctural transition occurs at $T_N = 203$ K, evidenced by magnetometry, heat capacity, and powder synchrotron X-ray experiments. This transition gives rise to long-range antiferromagnetic order with a rhombohedral-monoclinic symmetry change resulting from a slight distortion of the Co$_6$ octahedra. The antiferromagnetism is characterized by $\kappa_M = (00\frac{1}{2})$, as revealed by powder neutron diffraction, and below $T = 20$ K a small uncompensated component with $\kappa_M = (000)$ emerges spontaneously and coexists with the antiferromagnetism. The Ising-like spins of Co$^{2+}$ reorient themselves below $T_N$ upon the application of a large magnetic field. This transition occurs at $H_C = 3.9$ T for $T = 180$ K and gives rise to a broad upturn in the $M$-$H$ curve at $T = 120$ K it sharpens into a kink that exhibits hysteresis. A “butterfly” loop is thus formed, with linear and reversible behavior at low fields and hysteresis loops at high fields. The loops at positive and negative fields in the $M$-$H$ merge beneath $T = 20$ K because $H_C$ decreases as temperature decreases. A latent heat is observed in temperature-dependent measurements and indicates that this metamagnetic transition is strongly first-order. The low-temperature $M$-$H$ trends toward saturation at $M_S = 1.0$ $\mu_B$/Co even though this is only one third the value expected for high-spin Co(II). This reduced $M_S$ points toward the ferrimagnetic nature of the field-induced state rather than a spin-state transition to a $S = 1/2$ ferromagnet. The structural and magnetic $H$-$T$ phase diagram is not typical of inorganic Co(II) compounds nor of other transition metal oxides.

V. ACKNOWLEDGMENTS

PTB is supported by the NSF Graduate Research Fellowship Program. RS and PTB acknowledge the support of the NSF through DMR 1105301. We acknowledge the use of MRL Central Facilities which are supported by the MRSEC Program of the NSF under Award No. DMR 1121053; a member of the NSF-funded Materials Research Facilities Network (www.mrfn.org). Use of data from the 11-BM beamline at the Advanced Photon Source was supported by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under
Contract No. DE-AC02-06CH11357. This work has benefited from the use of HIPD at the Lujan Center at the Los Alamos Neutron Science Center, funded by the DOE Office of Basic Energy Sciences. Los Alamos National Laboratory is operated by Los Alamos National Security LLC under DOE Contract DE-AC52-06NA25396.

* Electronic address: pbarton@mrl.ucsb.edu

1. B. Raveau and M. M. Seikh, *Cobalt Oxides: From Crystal Chemistry to Physics* (Wiley-VCH, 2012).
2. H. Kobayashi and T. Haseda, J. Phys. Soc. Jpn. **19**, 765 (1964).
3. J. R. Neilson, D. E. Morse, B. C. Melot, D. P. Shoemaker, J. A. Kurzman, and R. Seshadri, Phys. Rev. B **83**, 094418 (2011).
4. M. Kurmoo, Chem. Soc. Rev. **38**, 1353 (2009).
5. B. C. Melot, A. Goldman, L. E. Darago, J. D. Furman, E. E. Rodriguez, and R. Seshadri, J. Phys.: Cond. Matter. **22**, 506003 (2010).
6. D. Gatteschi, Adv. Mater. **6**, 635645 (1994).
7. E. Stryjewski and N. Giordano, Adv. Phys. **26**, 487 (1977).
8. R. L. Carlin and A. J. van Duyneveldt, Acc. Chem. Res. **13**, 231 (1980).
9. V. Tsurkan, S. Zherlitsyn, S. Yasin, V. Felea, Y. Skourski, J. Deisenhofer, H.-A. K. von Nidda, J. Wosnitza, and A. Loidl, Phys. Rev. Lett. **110**, 115502 (2013).
10. J. Barbier, Acta Cryst. **B45**, 201 (1989).
11. L. V. Udod, G. A. Petrakovskii, A. M. Vorotynov, O. A. Bayukov, D. A. Velikanov, A. V. Kartashov, A. F. Bovina, Y. G. Shvedenkov, M. Baran, and R. Szymczak, Physics of the Solid State **49**, 500 (2007).
12. H. M. Rietveld, J. Appl. Crystallogr. **2**, 65 (1969).
13. B. H. Toby, J. Appl. Crystallogr. **34**, 210 (2001).
14. H. M. Rietveld, J. Appl. Crystallogr. **37**, 724 (2004).
15. B. J. Campbell, H. T. Stokes, D. E. Tanner, and D. M. Hatch, J. Appl. Cryst. **39**, 607 (2006).
16. K. Momma and F. Izumi, J. Appl. Cryst. **41**, 653 (2008).
17. J. C. Lashley, R. Stevens, M. K. Crawford, J. Boerio-Goates, B. F. Woodfield, Y. Qiu, J. W. Lynn, P. A. Goddard, and R. A. Fisher, Phys. Rev. B **78**, 104406 (2008).
18. P. A. Joy and S. K. Date, J. Magn. Magn. Mater. **210**, 31 (2000).
19. K. Adachi, K. Sato, and M. Takeda, J. Phys. Soc. Jpn. **26**, 631 (1969).
20. M. M. Altarawneh, G.-W. Chern, N. Harrison, C. D. Batista, A. Uchida, M. Jaime, D. G. Rickel, S. A. Crooker, C. H. Mielke, J. B. Betts, et al., Phys. Rev. Lett. **109**, 037201 (2012).
21. G. J. Redhammer, A. Senyshyn, G. Tippelt, C. Pietzonka, G. Roth, and G. Amthauer, Phys. Chem. Minerals **37**, 311 (2010).
22. M. Matsuda, T. Hoshi, H. A. Katori, M. Kosaka, and H. Takagi, J. Phys. Soc. Jpn. **80**, 034708 (2011).
23. T. Hoshi, H. A. Katori, M. Kosaka, and H. Takagi, J. Magn. and Magn. Mater. **310**, e448 (2007).