Low-energy Ar\(^+\) and N\(^+\) ion beam induced chemical vapor deposition using hexamethyldisilazane for the formation of nitrogen containing SiC and carbon containing SiN films

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Abstract

We proposed an experimental methodology for producing films on substrates with an ion beam induced chemical vapor deposition (IBICVD) method using hexamethyldisilazane (HMDS) as a source material. In this study, both HMDS and ion beam were simultaneously injected onto a Si substrate. We selected Ar\(^+\) and N\(^+\) as the ion beam. The energy of the ion beam was 101 eV. Temperature of the Si substrate was set at 540 °C. After the experiments, films were found to be deposited on the substrates. The films were then analyzed by Fourier transform infrared (FTIR) spectroscopy, stylus profilometer, X-ray diffraction, atomic force microscopy, and X-ray photoelectron spectroscopy (XPS). The FTIR and XPS results showed that silicon carbide films containing small amount of nitrogen were formed when Ar\(^+\) ions were injected in conjunction with HMDS. On the other hand, in the cases of N\(^+\) ion beam irradiation, silicon nitride films involving small amount of carbon were formed. It was noted that no film deposition was observed when HMDS alone was supplied to the substrates without any ion beam injections.

Introduction

Silicon carbide (SiC) recently receives wide-spread attention of its superior characteristics and is used as a semiconductor material [1,2], while silicon nitride (SiN) has been similarly interested in its practical applicability in surface passivation [3]. In most cases, SiC films were produced with the chemical vapor deposition (CVD) method using silane and propane as source gas [4]. On the other hand, SiN films could be produced using silane and ammonia [5].

An ion beam induced chemical vapor deposition (IBICVD) method has long been recognized to be useful for the preparation of various oxide films such as TiO\(_2\) [6,7], Al\(_2\)O\(_3\) [7], and
It is conceivable that the practical application of the IBICVD method to topics other than the oxide film formation would be important and significant subjects. At present, the IBICVD method is attracting much attention for the fabrications of three-dimensional nanostructures [9] and ferromagnetic materials [10].

Recently, the IBICVD method is found to be of use for the formation of SiC films. Matsutani et al. used hexamethyldisilane \[(\text{CH}_3)_3\text{SiSi(CH}_3)_3\] as a source material of IBICVD experiments [11]. In their experiments, Ar ion beams were injected to Si substrates in conjunction with the gas flow of hexamethyldisilane. They found that SiC films were formed on the substrates due to the dissociation of hexamethyldisilane by the Ar ion impact. Matsutani et al. also confirmed that silicon dioxide films could be deposited with the IBICVD method using hexamethyldisiloxane \[(\text{CH}_3)_3\text{SiOSi(CH}_3)_3\] [12]. Subsequently, Yoshimura et al. succeeded to deposit SiC films with the IBICVD method using methylsilane [13].

Meanwhile, hexamethyldisilazane [HMDS, \((\text{CH}_3)_3\text{SiNHSi(CH}_3)_3\)] has long been used in various scientific and technological studies. For examples, HMDS is available for the preparation of various biological specimens for microscopy [14,15], chemical syntheses [16–20], and the formation of SiN [21–23] and silicon carbonitride (SiCN) [24–30] films. HMDS has also been shown to be available for the deposition of SiC [31,32] and diamond-like carbon (DLC) [33].

The HMDS molecule has a chemical structure which is similar to hexamethyldisilane and hexamethyldisiloxane. Nevertheless, film deposition experiments with the IBICVD method using HMDS have not been carried out yet.

In this study, we tried to deposit films with the IBICVD method by injecting both ion beam and HMDS to a Si substrate. Both Ar\(^+\) and N\(^+\) were selected as the injecting ions. We examined the mass and energy distributions of the injecting ions prior to the IBICVD film deposition experiments, and after the IBICVD experiments, we assessed deposited films by Fourier transform infrared (FTIR) spectroscopy, stylus profilometer, X-ray diffraction (XRD), atomic force microscopy (AFM), and X-ray photoelectron spectroscopy (XPS).

**Materials and methods**

This study was carried out with an ion beam system (ULVAC) [34]. The ion source of the system was a Bernas-type source [35]. In this study, pure Ar gas or pure N\(_2\) gas was used as a source gas for the Ar\(^+\) or N\(^+\) ion beam production. Ar\(^+\) or N\(^+\) ions produced in the ion source were extracted and accelerated by a high voltage of -15 kV. Then, the ions were mass-selected by a mass selector to remove impurity ions. The mass-selected Ar\(^+\) or N\(^+\) ion beams were finally decelerated to 100 eV and then led to the processing chamber. The base vacuum in the processing chamber is approximately 1x10\(^{-6}\) Pa. The mass and energy of the ion beams can be determined by a plasma process monitor PPM-421 (balzers).

A schematic diagram of the IBICVD equipment in the processing chamber of the ion beam system is shown in Fig. 1. An untreated Si substrate (15x15 mm) was set on the substrate holder in the processing chamber (Fig 1). The energy of the ion beam was set at a low level (100 eV) to avoid significant damages of the deposited films by the ion injection. The incident angle of the ion beam was set to be normal to the substrate surface. Liquid HMDS in a container was bubbled using Ar gas and the resulting mixed gas of HMDS and Ar was supplied onto the substrate surface at a flow rate of 0.7 sccm through a stainless-steel tube connected to a mass flow controller (Fig 1). The gas pressure in the processing chamber was 1x10\(^{-3}\) Pa during the deposition experiment. The temperature of the Si substrate during the deposition experiment was set at 540 °C.
After the IBICVD experiments, we assessed deposited films by FTIR spectroscopy (FT/IR-410, Jasco), stylus profilometer (P-15, KLA-Tencor), XRD (RINT2200, RIGAKU), AFM (JSTM-4200D, Jeol), and XPS (AXIS-165x, Kratos).

Results

Production of Ar\(^+\) and N\(^+\) ion beams

Before the IBICVD film deposition experiments, the mass spectrum of the ion beam was measured with PPM-421. It was found that only a single peak appeared at the mass number of 40 [Fig 2a] or 14 [Fig 2b], suggesting the presence of pure Ar\(^+\) or N\(^+\) ions without impurity ions.

The ion energy spectra of Ar\(^+\) and N\(^+\) ion beams were also measured with PPM-421, as shown in Fig 3. Fig 3a and 3b show that the peak energies of (a) Ar\(^+\) and (b) N\(^+\) ion beams...
were 101 eV. Fig 3a and 3b also show that the ion beams used in this study were nearly monochromatic in energy distribution.

The profiles of current intensities for (a) Ar$^+$ and (b) N$^+$ ion beams were measured using a molybdenum orifice plate. The orifice diameter was 2 mm. The profiles are shown in Fig 4a and 4b. The full widths at half maximum of the profiles were about (a) 7 mm and [Fig 4a] and (b) 5 mm [Fig 4b].

Fig 3. The energy spectra of (a) Ar$^+$ and (b) N$^+$ ion beams.

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Fig 4. Typical intensity profiles for (a) Ar$^+$ and (b) N$^+$ ion beams. The ion energies of Ar$^+$ and N$^+$ ion beams were set at 100 eV. Horizontal axis represents the distance in the vertical direction.

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Supply of hexamethyldisilazane to Si substrates with and without ion beams

Then, we prepared three samples, (a), (b), and (c). For the sample (a), a Si substrate was supplied with HMDS gas flow in conjunction with Ar\(^+\) ion beams. On the other hand, both HMDS and N\(^+\) ions were injected to a Si substrate for the sample (b). For the sample (c), a Si substrate was supplied with HMDS without any ion beams. For the samples (a)-(c), the duration of the experiments was 4h.

We obtained FTIR spectra of these three samples in a transmission method using an FTIR spectrometer FT/IR-410 after the subtraction of the FTIR spectrum due to the Si substrate. The FTIR spectra of the samples (a) and (b) are shown in Fig 5a and 5b. The spectrum in Fig 5a has an absorption band at about 790 cm\(^{-1}\), showing that SiC film was deposited on the substrate [36]. On the other hand, the spectrum in Fig 5b has an absorption band due to Si-N [22], clearly indicating that SiN film was formed on the substrate. The FTIR spectrum of the sample (c) is shown in Fig 5c. On the contrary, Fig 5c shows no distinct peak, showing that almost no film formation occurred. We have also confirmed that no film formation was observed when substrates were irradiated with ion beams without HMDS. These results suggest that both of ion beam (Ar\(^+\) or N\(^+\)) and HMDS were necessary for the formation of films on substrates. It was also noted that the peaks at about 1500 and 2400 cm\(^{-1}\) in Fig 5a–5c
were attributed to water and carbon dioxide [37] in the ambient air inside the FT/IR-410 chamber.

**Low-energy Ar$^+$ and N$^+$ ion beam induced chemical vapor deposition using hexamethyldisilazane**

The thicknesses of the films on the samples (a) and (b) were too thin to measure with a stylus profiler. Therefore, we prepared two new samples. In this case, the duration of the deposition experiment was 28 h, which was 7 times larger than the previous experiments. From now on, we will refer to these two samples as films (a) and (b). For the film (a), a Si substrate was injected with HMDS in conjunction with Ar$^+$ ion beams. For the film (b), both of HMDS and N$^+$ ions were injected to a Si substrate. The average currents of (a) Ar$^+$ and (b) N$^+$ ions were 1.4 and 2.0 $\mu$A, respectively. The total Ar$^+$ and N$^+$ ion doses were (a) 0.9x10$^{18}$ and (b) 1.3x10$^{18}$ ions.

Firstly, we measured the thicknesses of the films (a) and (b) with a stylus profiler P-15. The measurement showed that the thicknesses of the films (a) and (b) were about 20 and 200 nm, respectively. Although the doses of injected Ar$^+$ and N$^+$ ions were similar in the films (a) and (b), the film (b) was much thicker than the film (a).

Then, we obtained FTIR spectra of these two films. The FTIR spectra of the films (a) and (b) are shown in Fig 6a and 6b. The spectra in Fig 6a and 6b are similar to those in Fig 5a and 5b, respectively.

Subsequently, we analyzed the films (a) and (b) with an X-ray diffractometer RINT2200 using K$_{\alpha}$1 of Co. The XRD patterns (θ-2θ method) of the films (a) and (b) are shown in Fig 7a and 7b. No distinct peaks were observed in both Fig 7a and 7b, suggesting that there were no crystalline structures in the films (a) and (b) and both films were amorphous.

Then, the films (a) and (b) were measured with JSTM-4200D. Fig 8a and 8b correspond to AFM images of the films (a) and (b). The root mean square surface roughness values for the films (a) and (b) were (a) 4 nm and (b) 14 nm, respectively.

![Fig 6. The Fourier-transform infrared spectra of films deposited on Si substrates following the injection of (a) Ar$^+$ and (b) N$^+$ ions in conjunction with hexamethyldisilazane. Substrate temperature during the deposition experiments was 340 °C. The duration of the deposition experiments was 28 h.](https://doi.org/10.1371/journal.pone.0259216.g006)
Finally, the films (a) and (b) were analyzed with an XPS instrument AXIS-165x. Before the measurement, the films were etched with an Ar beam for removing contaminants. Fig 9a and 9b show the XPS spectra in the Si 2p region of the films (a) and (b). The peak of the binding energy in Fig 9a corresponds to that of SiC [38]. On the other hand, the peak binding energy in Fig 9b agrees with the presence of SiN [39].

An evaluation of the XPS data from the film (a) showed that the C to Si atomic ratio was 1.0. The XPS data also showed that a small amount of nitrogen atoms was included in the film (a). The nitrogen concentration was estimated to be 6%. On the other hand, the atomic concentration ratio of N to Si for the film (b) was evaluated to be 0.9. The XPS data showed that carbon atoms were included in the film (b). The carbon concentration of the film was estimated to be 10%.

Discussion

In this study, FTIR and XPS assays of the film (a) showed that the nitrogen containing SiC film was formed on the substrate surface when Ar⁺ ions were injected in conjunction with HMDS. In previous papers, it was shown that nitrogen containing SiC films were available for thin overcoats for flexible media [40], materials with low dielectric constant [41], and materials of passivation layer [42].

We also found that the carbon containing SiN film was formed on the substrate when N⁺ ions were injected in conjunction with HMDS. There have been many papers to focus on SiN and SiCN film depositions with various experimental methods using HMDS [21–30]. These papers showed that the atomic concentration of Si, C, and N of films deposited using HMDS had a strong correlation with the experimental conditions. For example, it is necessary for the SiN deposition to investigate an optimum experimental condition, which can minimize the carbon elements to be incorporated in the film. The atomic concentration of carbon in the
film (b) in this study seems to be reduced by increasing the dose of N$^+$ ions injected. However, it is difficult to further increase the N$^+$ ion beam intensity in our ion beam system.

SiN films can also be deposited on substrates using hexamethyldisiloxane and ammonia [43,44]. Ammonia should be, however, handled carefully in these deposition experiments because of being extremely harmful. On the other hand, the method presented in this paper has an advantage over other ordinary methods, all of which use ammonia, because both HMDS and nitrogen are less harmful.

**Conclusion**

We here proposed an experimental methodology which made it possible to form films on substrates with an IBICVD method using HMDS.
We prepared three samples, (a), (b), and (c). For the sample (a), a Si substrate was supplied with HMDS gas flow in conjunction with Ar$^+$ ion beams. On the other hand, both HMDS and N$^+$ ions were simultaneously injected to a Si substrate for producing the sample (b). The sample (c) was produced on a Si substrate by providing HMDS without any ion beams. The Ar$^+$ and N$^+$ ion energies were both 101 eV. The temperature of the Si substrates was set at 540 ºC. The duration of the experiments was 4h. After the experiments, the FTIR spectra of the samples (a) and (b) suggested that (a) SiC and (b) SiN films were actually deposited on the substrates. The FTIR spectrum of the sample (c), however, showed no distinctive peak, suggesting that no film deposition was observed when HMDS alone was supplied.

The thicknesses of the films on the samples (a) and (b) were too thin. Therefore, we prepared additional two new samples, films (a') and (b'). In these cases, the duration of the deposition experiments was 28 h. For the film (a'), a Si substrate was injected with HMDS in conjunction with Ar$^+$ ion beams. For the film (b'), both of HMDS and N$^+$ ions were simultaneously injected to a Si substrate. The thicknesses of the films (a') and (b') were 20 and 200 nm, respectively. The FTIR spectra of the films (a') and (b') were similar to those of the samples (a) and (b), respectively. The XPS results of the film (a') showed that a SiC film containing small amount of nitrogen was formed when Ar$^+$ ions were injected in conjunction with HMDS. On the other hand, the XPS results of the film (b') indicated that a SiN film involving carbon was formed.

We conclude that the IBICVD method using HMDS is useful for film formations.

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