Charge transfer across transition metal oxide interfaces: emergent conductance and new electronic structure

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Abstract

We perform density functional theory plus dynamical mean field theory calculations to investigate internal charge transfer in an artificial superlattice composed of alternating layers of vanadate and manganite perovskite and Ruddlesden-Popper structure materials. We show that the electronegativity difference between vanadium and manganese causes moderate charge transfer from VO\textsubscript{2} to MnO\textsubscript{2} layers in both perovskite and Ruddlesden-Popper based superlattices, leading to hole doping of the VO\textsubscript{2} layer and electron doping of the MnO\textsubscript{2} layer. Comparison of the perovskite and Ruddlesden-Popper based heterostructures provides insights into the role of the apical oxygen. Our first principles simulations demonstrate that the combination of internal charge transfer and quantum confinement provided by heterostructuring is a powerful approach to engineering electronic structure and tailoring correlation effects in transition metal oxides.
I. INTRODUCTION

Advances in thin film epitaxy growth techniques have made it possible to induce emergent electronic [1–10], magnetic [11–13] and orbital [14, 15] states, which are not naturally occurring in bulk constituents, at atomically sharp transition metal oxide interfaces [16–19]. For example, the interface between the two nonmagnetic band insulators LaAlO$_3$ and SrTiO$_3$ [4] has been reported to exhibit both conductance [20] and magnetism [21] (see reviews [22–25] and references therein).

In LaAlO$_3$/SrTiO$_3$ and related heterostructures, the interface electron gas is believed to be produced by the polar catastrophe mechanism, which leads to the transfer of charge from the sample surface to the interface. Here, we consider a different mechanism for controlling the electronic properties of an interface: namely, electronegativity-driven charge transfer. Recently, we have shown that internal charge transfer in a LaTiO$_3$/LaNiO$_3$ superlattice transforms metallic LaNiO$_3$ into a $S = 1$ Mott insulator and Mott insulating LaTiO$_3$ into a $S = 0$ band insulator [26]. A natural question arises: can we reverse the process and utilize internal charge transfer to induce conductance via oxide interfaces? In this regard, it is very tempting to explore Mott interfaces (one or both constituents are Mott insulators) due to the unusual phenomena (colossal magnetoresistance and high temperature superconductivity) exhibited in certain doped Mott insulators.

In this paper we use density functional theory + dynamical mean field theory (DFT+DMFT) to theoretically design a superlattice with emergent metallic behavior. We explore two different types of structure: the perovskite structure (referred to as 113-type) and the $n = 1$ Ruddlesden-Popper structure (referred to as 214-type). Among the four bulk constituents (SrVO$_3$, SrMnO$_3$, Sr$_2$VO$_4$ and Sr$_2$MnO$_4$), all are correlation-driven insulators except SrVO$_3$, which is a moderately correlated metal [27–29]. We show that the difference of electronegativity between the elements V and Mn drives internal charge transfer from V to Mn sites, leading to a “self-doping” at the interface and possibly inducing conductance as Mn sites become weakly electron doped and V sites hole doped.

The rest of the paper is organized as follows: Section II presents the theoretical methods. A schematic of band alignment is presented in Section III to illustrate the underlying mechanism of charge transfer. All the bulk results from ab initio calculations are in Section IV and the results of vanadate-manganite superlattices are in Section V, both of which provide qual-
FIG. 1: Simulation cells of A) bulk SrVO₃, B) bulk SrMnO₃ and C) SrVO₃/SrMnO₃ superlattice; D) bulk Sr₂VO₄, E) bulk Sr₂MnO₄ and F) Sr₂VO₄/Sr₂MnO₄ superlattice. The green atoms are Sr. The blue and purple cages are VO₆ and MnO₆ octahedra, respectively. The stacking direction of the superlattice is the [001] axis.

The computations support and quantitative corrections to the schematic. The conclusions are in Section VI. Four Appendices present technical details relating to the insulating gaps of Sr₂VO₄ and Sr₂MnO₄, alternative forms of the double counting correction, and the possibility of two consecutive repeating layers (i.e. 2/2 superlattices instead of 1/1 superlattices).

II. COMPUTATIONAL DETAILS

The DFT calculations of our DFT+DMFT calculations is performed using a plane-wave basis, as implemented in the Vienna Ab-initio Simulation Package (VASP) using the Projector Augmented Wave (PAW) approach. The correlated subspace and the orbitals with which it mixes are constructed using maximally localized Wannier functions defined over the full 10 eV range spanned by the p-d band.
complex, resulting in a well localized set of $d$-like orbitals. We approximately find the stationary point of our DFT+DMFT functional by first converging the charge density within DFT and then subsequently converging the DMFT equations. This procedure is found to yield reasonable results in calculations in bulk systems \cite{29, 42, 43}.

For the bulk materials, we consider two structures: the experimental one and the theoretical relaxed structure obtained by the use of DFT within the local density approximation (LDA) \cite{44}. For the superlattice, we use DFT-LDA to obtain the relaxed structure. The simulation cell is illustrated in Fig. 11. The stacking direction of the superlattice is along [001]. We use an energy cutoff 600 eV. A $12 \times 12 \times 12 \left( \frac{L_x}{L_z} \right)$ ($L_x$ and $L_z$ are the lattice constants along the x and z directions, and $[x]$ is the integer part of $x$) Monkhorst-Pack grid is used to sample the Brillouin zone. Both cell and internal coordinates are fully relaxed until each force component is smaller than 10 meV/Å and the stress tensor is smaller than 10 kBar. A higher energy cutoff, a denser $k$-point sampling, and a smaller force threshold are employed to test the convergence and no significant changes are found in the results.

For the vanadates, we treat the empty $e_g$ orbitals with a static Hartree-Fock approximation (recent work shows this approximation is adequate to describe the electronic structure of vanadates \cite{29}), while correlations in the V $t_{2g}$ manifold are treated within single-site DMFT including the Slater-Kanamori interactions using $U_V = 5$ eV and $J_V = 0.65$ eV \cite{45, 46}. For manganites, we treat the correlations on all the five Mn $d$ orbitals within single-site DMFT using the Slater-Kanamori interactions with $U_{Mn} = 5$ eV and $J_{Mn} = 1$ eV \cite{47}. The DMFT impurity problem is solved using the continuous time quantum Monte Carlo method \cite{48, 49}. In order to use the “segment” algorithm \cite{50}, we neglect the exchange and pairing terms in the Slater-Kanamori Hamiltonian. All the calculations are paramagnetic and the temperature is set to 232 K.

An important outstanding issue in the DFT+DMFT procedure is the “double counting correction” which accounts for the part of the Slater-Kanamori interactions already included in the underlying DFT calculation and plays an important role by setting the mean energy difference between the $d$ and $p$ bands. The $p$-$d$ separation plays a crucial role in determining the band alignment, which affects the charge transfer. However, currently there is no exact procedure for the double counting correction. We use the $U'$ double counting method recently introduced by Park and two of us \cite{43}, where the parameter $U'$ is the prefactor in the double-counting which determines the $p$-$d$ separation and equivalently the number of electrons in
the $d$-manifold. In this study, $U'$ is chosen to produce an energy separation between the O $p$ and transition metal $d$ bands which is consistent with photoemission experiments. Our main qualitative conclusions do not depend on the details of the double counting scheme; in particular we show here they hold also for the conventional fully localized limit (FLL) double counting \[51\] which is the $U' = U$ limit of the method of Ref. \[43\].

The spectral function presented throughout this work is defined as follows:

$$A_i(\omega) = -\frac{1}{\pi N_k} \sum_k \text{Im} \left( \left[ (\omega + \mu)\mathbf{I} - H_0(k) - \Sigma(\omega) + V_{dc} \right]^{-1}\right)_{ii}$$  \hspace{1cm} (1)$$

where $i$ is the label of a Wannier function, $N_k$ is the number of $k$-points, $\mathbf{I}$ is an identity matrix, $H_0(k)$ is the DFT-LDA band Hamiltonian in the matrix form using the Wannier basis. $\Sigma(\omega)$ is the self-energy and is understood as a diagonal matrix only with nonzero entries on the correlated orbitals. Local tetragonal point symmetry of the V and Mn sites ensures that $\Sigma(\omega)$ is diagonal within the correlated orbital subspace. $V_{dc}$ is the double counting potential. $\mu$ is the chemical potential.
III. SCHEMATIC OF BAND STRUCTURE AND BAND ALIGNMENT

We consider the following materials as components of the superlattice: SrVO$_3$, a moderately correlated metal with nominal $d$-valence $d^1$; Sr$_2$VO$_4$, a correlation-driven insulator also with nominal valence $d^1$; and SrMnO$_3$ and Sr$_2$MnO$_4$, both of which are $d^3$ correlation-driven (Mott) insulators. Fig. 1 shows the atomic structure of the bulk phases of the constituent materials and the corresponding superlattices. Fig. 1A, B, and C are bulk SrVO$_3$, bulk SrMnO$_3$, and SrVO$_3$/SrMnO$_3$ superlattice, respectively. Fig. 1D, E, and F are bulk Sr$_2$VO$_4$, bulk Sr$_2$MnO$_4$, and Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice, respectively. In both superlattices, the stacking direction is along the [001] axis. In the 214-type, the V atoms are shifted by a $(\frac{1}{2}, \frac{1}{2})$ lattice constant in the $xy$ plane relative to the Mn atoms.

Fig. 2 is a schematic of the band structure of bulk vanadates, bulk manganites, and the band alignments in the superlattice (the small insulating gap of Sr$_2$VO$_4$ is not relevant here. There is a large energy separation (around 2 eV) between V $d$ and O $p$ states (see Fig. 2A). In the Mn-based materials (see Fig. 2B), the highest occupied states are Mn $t_{2g}$-derived and the lowest unoccupied states are Mn $e_g$-derived. Due to the electronegativity difference between V and Mn, visible as the difference in the energy separation of the transition metal $d$ levels from the oxygen $p$ levels, if we align the O $p$ states between vanadates and manganites (see Fig. 2C), the occupied V $t_{2g}$ states overlap in energy with the unoccupied Mn $e_g$ states. The overlap drives electrons from V sites to Mn sites. As the superlattice is formed, a common Fermi level appears across the interface and thus we expect that Mn $e_g$ states become electron doped and V $t_{2g}$ states hole doped.

We make two additional points: i) though SrVO$_3$ is a metal and Sr$_2$VO$_4$ is an insulator with a small energy gap (around 0.2 eV) [27], the near Fermi level electronic structure does not affect the band alignment and therefore the internal charge transfer is expected to occur no matter whether there is a small energy gap in V $t_{2g}$ states at the Fermi level or not; ii) in our schematic, we assume that the main peak of O $p$ states are exactly aligned between the vanadates and manganites in the superlattices. Of course, real material effects will spoil any exact alignment. We will use $ab$ initio calculations to provide quantitative information on how O $p$ states are aligned between the two materials.
TABLE I: The in-plane and out-of-plane V-O and Mn-O bond lengths $l$ of SrVO$_3$, SrMnO$_3$, Sr$_2$VO$_4$ and Sr$_2$MnO$_4$. The corresponding VO$_6$ and MnO$_6$ octahedral volumes $\Omega$ are also calculated. The relaxed structures are obtained from DFT-LDA non-spin-polarized calculations. The values in parentheses are experimental values which are referenced in the main text.

|                  | SrVO$_3$          | SrMnO$_3$       | SrVO$_3$/SrMnO$_3$ |
|------------------|-------------------|-----------------|-------------------|
| $l_{\text{in}}$(V-O) | 1.89 Å (1.92 Å)   | –               | 1.88 Å            |
| $l_{\text{out}}$(V-O) | 1.89 Å (1.92 Å)   | –               | 1.85 Å            |
| $\Omega_{\text{VO}_6}$ | 9.00 Å$^3$ (9.44 Å$^3$) | –               | 8.72 Å$^3$        |
| $l_{\text{in}}$(Mn-O) | –                | 1.86 Å (1.90 Å) | 1.88 Å            |
| $l_{\text{out}}$(Mn-O) | –                | 1.86 Å (1.90 Å) | 1.89 Å            |
| $\Omega_{\text{MnO}_6}$ | –                | 8.58 Å$^3$ (9.15 Å$^3$) | 8.91 Å$^3$        |

|                  | Sr$_2$VO$_4$      | Sr$_2$MnO$_4$   | Sr$_2$VO$_4$/Sr$_2$MnO$_4$ |
|------------------|-------------------|-----------------|-----------------------------|
| $l_{\text{in}}$(V-O) | 1.88 Å (1.91 Å)   | –               | 1.85 Å                      |
| $l_{\text{out}}$(V-O) | 1.96 Å (1.95 Å)   | –               | 1.93 Å                      |
| $\Omega_{\text{VO}_6}$ | 9.24 Å$^3$ (9.49 Å$^3$) | –               | 8.81 Å$^3$                  |
| $l_{\text{in}}$(Mn-O) | –                | 1.82 Å (1.90 Å) | 1.85 Å                      |
| $l_{\text{out}}$(Mn-O) | –                | 1.99 Å (1.95 Å) | 1.99 Å                      |
| $\Omega_{\text{MnO}_6}$ | –                | 8.79 Å$^3$ (9.39 Å$^3$) | 9.08 Å$^3$                  |

IV. BULK PROPERTIES

This section is devoted to properties of vanadates and manganites in their bulk single crystalline form. We perform DFT+DMFT calculations on both experimental structures and relaxed atomic structures obtained from DFT-LDA. The DFT-LDA relaxed V-O and Mn-O bond lengths, as well as the volume of VO$_6$ and MnO$_6$ octahedra, are summarized in Table I along with the experimental bond lengths and octahedral volumes (in parentheses) for comparison. However, in order to directly compare to the photoemission data, we only present the spectral functions that are calculated using the experimental structures.
FIG. 3: Orbitally resolved spectral function of A) SrVO$_3$ and B) Sr$_2$VO$_4$; C) SrMnO$_3$ and D) Sr$_2$MnO$_4$. The pink dots are the experimental spectra for either SrVO$_3$ or SrMnO$_3$ (identical data are plotted alongside the theoretical spectra for the Ruddlesden-Popper structures) [52]. For vanadates, $U'$ double counting is employed with $U_V = 5$ eV and $U'_V = 3.5$ eV. The red (very thick), blue (thin) and green (thick) curves are V $t_{2g}$, V $e_g$ and O $p$ projected spectral functions, respectively. For manganites, $U'$ double counting is employed with $U_{Mn} = 5$ eV and $U'_{Mn} = 4.5$ eV. The red (thin), blue (very thick) and green (thick) curves are Mn $t_{2g}$, Mn $e_g$ and O $p$ projected spectral functions, respectively. The Fermi level is set at zero energy.

A. Bulk vanadates

We begin with bulk vanadates: SrVO$_3$ and Sr$_2$VO$_4$. SrVO$_3$ has a cubic structure with a lattice constant $a = 3.841$ Å [53]. Sr$_2$VO$_4$ forms $n = 1$ Ruddlesden-Popper structure with the in-plane lattice constant $a = 3.826$ Å and the out-of-plane lattice constant $c = 12.531$ Å [54]. We use a Hubbard $U_V = 5$ eV on both vanadate materials to include correlation effects on V $d$ orbitals, which is in the vicinity of previous studies [45, 46].
Fig. 3 shows the orbitally-resolved spectral function \( A(\omega) \) of bulk SrVO\(_3\) (Fig. 3A) and bulk Sr\(_2\)VO\(_4\) (Fig. 3B), along with the experimental photoemission data for bulk SrVO\(_3\) [52]. The threshold of \( \text{O} \) \( p \) states is around 2 eV below the Fermi level. We find that \( U'_V = 3.5 \) eV yields a reasonable agreement between the calculated \( \text{O} \) \( p \) states and experimental photoemission data. At \( U_V = 5 \) eV, with the \( p-d \) separation fixed by the experimental photoemission data, our DFT+DMFT calculations find SrVO\(_3\) to be metallic, consistent with the experiment. However, they do not reproduce a Mott insulating state in Sr\(_2\)VO\(_4\), as observed in experiment. We show in the Appendix A that a metal-insulator transition does occur in Sr\(_2\)VO\(_4\) with an increasing Hubbard \( U_V \) and a fixed \( p-d \) separation (via \( U'_V \)). However, the critical \( U_V \) is larger than typical values employed previously in literature for the vanadates [45, 46]. It is possible that the experimentally observed narrow-gap insulating behavior (experimentally observed to persist above the Néel temperature [27, 55]) arises from long-range magnetic correlations and spatial correlations that are not captured in our single-site paramagnetic DMFT calculation. These correlations relate to low energy scale physics [56] and are not expected to affect the charge transfer energetics of interest here.

### B. Bulk manganites

Next we discuss the bulk manganites: SrMnO\(_3\) and Sr\(_2\)MnO\(_4\). For ease of comparison with the superlattice results to be shown in the next section, we study here the cubic phase of SrMnO\(_3\) (isostructural to SrVO\(_3\)) with the lattice constant of \( a = 3.801 \) Å (though other structures of SrMnO\(_3\) also co-exist) [57]. Sr\(_2\)MnO\(_4\) forms the \( n = 1 \) Ruddlesden-Popper structure with in-plane and out-of-plane lattice constants \( a = 3.802 \) Å and \( c = 12.519 \) Å [58]. Consistent with the experimental estimation of Hubbard \( U \) from photoemission data [47], we use a Hubbard \( U_{\text{Mn}} = 5 \) eV on both materials to include correlation effects on Mn \( d \) orbitals.

Fig. 3 shows the orbitally-resolved spectral function \( A(\omega) \) of bulk SrMnO\(_3\) (Fig. 3C) and Sr\(_2\)MnO\(_4\) (Fig. 3D) [59]. The threshold of \( \text{O} \) \( p \) states is around 1 eV below the Fermi level. We find that \( U'_{\text{Mn}} = 4.5 \) eV provides a good agreement between the calculated \( \text{O} \) \( p \) states and experimental photoemission data. We observe that for these parameters the occupied Mn \( t_{2g} \) states are visible as a peak slightly above the leading edge of the oxygen band. We will show in Appendix B that modest changes of parameters will move this peak
slightly down in energy so that it merges with the leading edge of the oxygen $p$ states. The experimental situation is not completely clear. Published x-ray photoelectron spectroscopy work $[60, 61]$ indicates a resolvable $t_{2g}$ peak at or slightly above the leading edge of the oxygen bands; other studies including recent photoemission measurements $[52, 62]$ do not find a separately resolved $t_{2g}$ peak. The issue is not important for the results of this paper but further investigation of the location of the $t_{2g}$ states would be of interest as a way to refine our knowledge of the electronic structure of the manganites. With this value of $U'_{\text{Mn}}$, the theory produces a small energy gap around 0.5 eV in both SrMnO$_3$ and Sr$_2$MnO$_4$. However, the gap value is $U_{\text{Mn}}$-dependent. We show in Appendix B that with the $p$-$d$ separation fixed, via the adjustment of $U'_{\text{Mn}}$, a larger $U_{\text{Mn}}$ increases the Mott gap by further separating the Mn lower and upper Hubbard bands. However, for the value of $U_{\text{Mn}}$ (around 5 eV) that is extracted from photoemission experiments $[47]$, the size of the Mott gap of Sr$_2$MnO$_4$ is substantially underestimated, compared to the optical gap (around 2 eV) in experiment $[27]$. This discrepancy may arise because this calculation does not take into account spatial correlation $[63]$. However, the Mott gap is separated by Mn $t_{2g}$ and $e_g$ states, while the energy difference between O $p$ states and Mn $e_g$ states (i.e. $p$-$d$ separation) is fixed by the experimental photoemission data (via $U'_{\text{Mn}}$). We will show in the next section as well as in the Appendix B that it is the $p$-$d$ separation that controls the charge transfer and therefore the underestimation of the Mott gap does not significantly affect our main results.

V. VANADATE-MANGANITE SUPERLATTICES

In this section we discuss vanadate-manganite superlattices. There are two types: we refer to SrVO$_3$/SrMnO$_3$ superlattice as 113-type and refer to Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice as 214-type. The two types of superlattices have similarities and differences. In both types, the charge transfer from V sites to Mn sites occurs, in which electron dopes the Mn $e_g$ states and drains the V $t_{2g}$ states at the Fermi level. However, in the 214 type, the VO$_6$ and MnO$_6$ octahedra are decoupled and the charge transfer arises mainly from the electronegativity difference between V and Mn elements. In the 113 type, in addition to the electronegativity difference between V and Mn, the movement of the shared apical oxygen changes the hybridization and thus also affects the charge transfer. We will show below that due to the movement of the shared apical oxygen atom, the 113-type superlattice generically has a
more enhanced charge transfer than the 214-type superlattice.

We discuss the phenomena of charge transfer in terms of: 1) structural properties, 2) electronic properties and 3) direct electron counting.

### A. Structural properties

Table I shows the DFT-LDA relaxed structure of SrVO\textsubscript{3}/SrMnO\textsubscript{3} and Sr\textsubscript{2}VO\textsubscript{4}/Sr\textsubscript{2}MnO\textsubscript{4} superlattices as well as the bulk materials. We see that the VO\textsubscript{6} octahedron is smaller in the superlattice than in the bulk, while the MnO\textsubscript{6} octahedron is larger. This is suggestive that the VO\textsubscript{6} octahedron loses electrons and the MnO\textsubscript{6} octahedron gains electrons (i.e that internal charge transfer from V to Mn sites occurs), and this will be quantified below.

### B. Electronic properties

Fig. 4 shows the orbitally resolved spectral function of the SrVO\textsubscript{3}/SrMnO\textsubscript{3} superlattice (left panels) and the Sr\textsubscript{2}VO\textsubscript{4}/Sr\textsubscript{2}MnO\textsubscript{4} superlattice (right panels). In both superlattices, the Mn \textit{e}\textsubscript{g} states emerge at the Fermi level, while in bulk manganites, there is a small gap in the Mn \textit{d} states (separated by Mn \textit{e}\textsubscript{g} and \textit{t}\textsubscript{2g}) in both materials. In the VO\textsubscript{2} layer, V \textit{t}\textsubscript{2g} states dominate at the Fermi level. Another feature worth noting is the O \textit{p} states of the MnO\textsubscript{2} and of the VO\textsubscript{2} layers. Though the very first peak of O \textit{p} states in the MnO\textsubscript{2} layer below the Fermi level is lined up with Mn \textit{t}\textsubscript{2g} states due to strong covalency, the main peak almost exactly overlaps with that of O \textit{p} states in the VO\textsubscript{2} layer. This supports our hypothesis in the schematic that the main peaks of O \textit{p} states of the VO\textsubscript{2} and MnO\textsubscript{2} layers are aligned in the superlattices. We need to mention that the general features in electronic structure of the superlattices are robust for different double counting schemes. We show in Appendix C that the standard FLL double counting yields a very similar electronic structure of the superlattices.

Next, we compare the V \textit{t}\textsubscript{2g} and Mn \textit{e}\textsubscript{g} states between the superlattices and bulk materials to show how the Fermi level shifts in the two constituents. Fig. 5A and B show the comparison of Mn \textit{e}\textsubscript{g} and O \textit{p} states of the MnO\textsubscript{2} layer between the superlattices and bulk manganites (A: 113-type and B: 214-type). The Fermi levels of bulk manganites and of the superlattices are lined up in the same figure.
FIG. 4: Orbitally resolved spectral function of vanadate-manganite superlattices. Left panels: SrVO$_3$/SrMnO$_3$ superlattice. Right panels: Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice. A) and B): Mn $t_{2g}$ (red thin) and Mn $e_g$ (blue thick) states; C) and D): V $t_{2g}$ (green thick) and V $e_g$ (violet thin) states; E) and F): O $p$ states of the MnO$_2$ layer (turquoise thick) and O $p$ states of the VO$_2$ layer (maroon thin). $U'$ double counting is employed with $U_V = U_{Mn} = 5$ eV and $U'_V = 3.5$ eV, $U'_{Mn} = 4.5$ eV. The Fermi level is set at zero point.

According to the schematic (Fig. 2), with respect to bulk manganites, both the Mn $d$ and O $p$ states in the MnO$_2$ layer are shifted towards the low energy-lying region due to the electron doping. Fig. 5A and B clearly reproduce this rigid shift in i) Mn $e_g$ states from the bulk (blue or thin dark curves) to the superlattice (red or thick light) and ii) in O $p$ states of the MnO$_2$ layer from the bulk (turquoise or thin light) to the superlattice (maroon or thick dark).

Similarly, Fig. 5C and D show the comparison of V $t_{2g}$ and O $p$ states (of the VO$_2$ layer) between the superlattices and bulk vanadates (C: 113-type and D: 214-type). The Fermi
FIG. 5: A) Comparison of Mn $e_g$ and O $p$ states of the MnO$_2$ layer between the SrVO$_3$/SrMnO$_3$ superlattice and bulk SrMnO$_3$. B) Comparison of Mn $e_g$ and O $p$ states of the MnO$_2$ layer between the Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice and bulk Sr$_2$MnO$_4$. C) Comparison of V $t_{2g}$ and O $p$ states of the VO$_2$ layer between the SrVO$_3$/SrMnO$_3$ superlattice and bulk SrVO$_3$. D) Comparison of V $t_{2g}$ and O $p$ states of the VO$_2$ layer between the Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice and bulk Sr$_2$VO$_4$. The Fermi level is set at zero energy. “SL” refers to the superlattices.

levels of bulk vanadates and of the superlattices are lined up in the same figure. According to the schematic (Fig. 2), since electrons are drained out of V $t_{2g}$ state, both the V $t_{2g}$ states and O $p$ states of the VO$_2$ layer are shifted towards the high energy-lying region, compared to their counterparts in bulk vanadates. This shift can be seen (Fig. 5C and D) i) in the V $t_{2g}$ states from the bulk (blue or thin dark curves) to the superlattice (red or thick light) and ii) in the O $p$ states of the VO$_2$ layer from the bulk (turquoise or thin light) to the superlattice (maroon or thick dark). However, since the peak of V $t_{2g}$ states at the Fermi level is much higher than that of Mn $e_g$ states, the shift in the V $t_{2g}$ states is much smaller than that in the Mn $e_g$ states. Fig. 5 reproduces our schematic of how V $t_{2g}$ and Mn $e_g$ states are shifted.
FIG. 6: Movement of the apical oxygen, corresponding changes in the energy of V $d$ and Mn $d$ states and enhancement of the charge transfer. A) SrVO$_3$/SrMnO$_3$ superlattice and B) Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice. The green atoms are Sr. The blue and purple cages are VO$_6$ and MnO$_6$ octahedra, respectively. The arrows on the oxygen atoms indicate the atom movement. The arrows on the metal $d$ states indicate the trend of energy shift.

and re-arranged to reach one common Fermi level in a vanadate-manganite superlattice. A possible consequence is electron (hole) conductance in the MnO$_2$ (VO$_2$) layer.

C. Direct electron counting

Now we calculate the occupancy on each orbital by performing the following integral:

$$N_i = \int_{-\infty}^{\infty} A_i(\omega)n_F(\omega)d\omega$$

where $A_i(\omega)$ is the spectral function for the $i$th orbital (defined from the Wannier construction), which is defined in Eq. (1). $n_F(\omega)$ is the fermion occupancy factor. In order to explicitly display the charge transfer phenomenon, we calculate the V $d$ and Mn $d$ occupancy in both bulk materials and the superlattices. We summarize the results in Table III.

We can see that $N_d(V)$ decreases and $N_d(Mn)$ increases from bulk to the superlattices and an average charge transfer from V to Mn is 0.40$e$ for the 113-type superlattice and 0.25$e$ for the 214-type superlattice. Moreover, due to the strong covalency between transition metal $d$ states and oxygen $p$ states, the occupancy of oxygen $p$ states also changes between bulk
materials and the superlattices. For this reason, the change in $d$ occupancy may not be an accurate representation of charge transfer.

We also calculate the total occupancy of VO$_2$ and MnO$_2$ layers and find that the total charge transfer between the two layers amounts to 0.53 for the 113-type superlattice and 0.38 for the 214-type superlattice. Unlike the 113-type superlattice in which the apical oxygen is shared by two octahedra, the 214-type superlattice has a unique property that each octahedron is decoupled between layers. Therefore in the superlattice, we can count the charge transfer from the VO$_6$ octahedron to the MnO$_6$ octahedron. Note that since we only take into account the $p$-$d$ band manifold, the V and Mn octahedra include all the Wannier states and therefore in bulk Sr$_2$VO$_4$, the number of electrons per VO$_4$ unit is exactly 25e and in bulk Sr$_2$MnO$_4$, the number of electrons per MnO$_4$ unit is exactly 27e. We find that relative to the bulk materials, the V octahedron of the 214-type superlattice loses 0.48e and Mn octahedron of the 214-type superlattice gains exactly 0.48e. Comparison of this 0.48e charge transfer to the 0.25e found by only considering $d$ orbitals further confirms that not only the transition metal $d$ states but also oxygen $p$ states participate in the charge transfer.

From Table II, we can see that the internal charge transfer is stronger in the 113-type superlattice, compared to the 214-type. We show below that the difference arises because in the 113-type superlattice, the apical oxygen is shared by the VO$_6$ and MnO$_6$ octahedra, whereas the octahedra are decoupled in the 214-type.

We see from Table II that due to the internal charge transfer, the VO$_6$ octahedron loses electrons and shrinks; on the other hand, the MnO$_6$ octahedron gains electrons and expands. Therefore the shared apical oxygen atom moves away from Mn sites and towards V sites (see Fig. 6A). A direct consequence is that the out-of-plane Mn-O hopping decreases and the out-of-plane V-O hopping increases. Since the V $d$ and Mn $d$ states are anti-bonding in nature, the changes in the metal-ligand hopping push the V $d$ states higher in energy and lower the energy of Mn $d$ states and thus enhance the internal charge transfer. In the 214-type superlattice, we have a different situation because the two oxygen octahedra have their own apical oxygen atoms, whose movements are decoupled. From the Table II the VO$_6$ shrinks and the apical oxygen atom of VO$_6$ moves towards the V atom, just like the 113-type superlattice. However, the MnO$_6$ expands but the apical oxygen does not move (the in-plane Mn-O bond does increase, so does the overall volume of MnO$_6$). Therefore, the energy of V $d$ states is increased due to the enhanced out-of-plane V-O hopping, but the
TABLE II: The occupancy of V \(d\) and Mn \(d\) states, as well as VO\(_2\) and MnO\(_2\) layers in vanadates, manganites and the superlattices. All the occupancies without the parentheses are calculated from Wannier basis using the DFT-LDA relaxed structures. The occupancies in the parentheses are calculated from Wannier basis using the experimental structures. \(\Delta N_d\) (\(\Delta N\)) is the average charge transfer between V \(d\) and Mn \(d\) states (VO\(_2\) and MnO\(_2\) layers, or VO\(_4\) and MnO\(_4\) octahedra), using the DFT-LDA relaxed structures.

|                  | SrVO\(_3\) | SrMnO\(_3\) | SrVO\(_3\)/SrMnO\(_3\) |
|------------------|------------|-------------|-------------------------|
| \(N_d(V)\)      | 2.09 (2.03)| 4.08 (4.06) | 1.73 4.51 0.40          |
| \(N(VO_2)\)     | 13.36 (13.34)| 15.36 (15.35)| 12.86 15.92 0.53    |
| \(\Delta N_d\)  |            |             |                         |

|                  | Sr\(_2\)VO\(_4\) | Sr\(_2\)MnO\(_4\) | Sr\(_2\)VO\(_4\)/Sr\(_2\)MnO\(_4\) |
|------------------|------------------|------------------|-------------------------|
| \(N_d(V)\)      | 2.07 (2.03)      | 4.12 (4.09)      | 1.86 4.41 0.25          |
| \(N(VO_2)\)     | 13.35 (13.36)    | 15.42 (15.44)    | 13.01 15.83 0.38        |

|                  | Sr\(_2\)VO\(_4\) | Sr\(_2\)MnO\(_4\) | Sr\(_2\)VO\(_4\)/Sr\(_2\)MnO\(_4\) |
|------------------|------------------|------------------|-------------------------|
| \(N(VO_4)\)     | 25.00 (25.00)    | 27.00 (27.00)    | 24.52 27.48 0.48        |

energy of Mn \(d\) states remains almost unchanged because the apical oxygen atom does not move (Fig. 6B). As a result, the charge transfer between V and Mn sites is weaker in the 214-type superlattice, compared to the 113-type superlattice.

Our discussions in this paper have focussed mainly on the \((\text{SrVO}_3)_1/(\text{SrMnO}_3)_1\) superlattice. Though \(m = 1\) superlattices (in the notation of \((\text{SrVO}_3)_m/(\text{SrMnO}_3)_m\)) are easy for theoretical studies, experimentally it is more practical to grow \(m = 2\) or larger \(m\) superlattices. We show in the Appendix D that comparing \((\text{SrVO}_3)_1/(\text{SrMnO}_3)_1\) and \((\text{SrVO}_3)_2/(\text{SrMnO}_3)_2\) superlattices, the charge transfer is very similar. However, for a large \(m\), we will have in-equivalent V sites and eventually the charge transfer will be confined to the interfacial region. Investigating the length scales associated with charge transfer is an important open question.
VI. CONCLUSIONS

We use DFT+DMFT calculations to show that due to the difference in electronegativity, internal charge transfer could occur between isostructural vanadates and manganites in both 113-type and 214-type superlattices. The charge transfer is enhanced by associated lattice distortions. The moderate electronegativity difference between Mn and V leads to moderate charge transfer, in contrast to the LaTiO$_3$/LaNiO$_3$ superlattice, in which a complete charge transfer fills up the holes on the oxygen atoms in the NiO$_2$ layer. The partially filled bands imply metallic conductance that could possibly be observed in transport, if the thin film quality is high enough that disorder is suppressed and Anderson localization does not occur [64]. Our study of a superlattice consisting of two different species of transition metal oxides establishes that internal charge transfer is a powerful tool to engineer electronic structure and tailor correlation effects in transition metal oxides. In particular, for vanadate-manganite superlattices, internal charge transfer may serve as an alternative approach to dope Mott insulators without introducing chemical disorder. Finally our work shows that in addition to perovskite structure, Ruddlesden-Popper structures can also be introduced in the design of oxide superlattices with tailored properties [65].

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Appendix A: Metal-insulator transition of Sr$_2$VO$_4$

In this appendix, we show that within single-site DMFT and with the $p$-$d$ separation fixed by the experimental photoemission data, there is a metal-insulator transition in Sr$_2$VO$_4$ with an increasing Hubbard $U_V$ ($U'_V$ is determined by the $p$-$d$ separation for each given $U_V$).
FIG. 7: Orbitally resolved spectral function of Sr$_2$VO$_4$. $U'$ double counting is employed with A) $U_V = 5$ eV, $U'_V = 3.5$ eV and B) with $U_V = 8$ eV, $U'_V = 6.8$ eV. The red (very thick), blue (thick) and green (thin) curves are V $t_{2g}$, V $e_g$ and O $p$ projected spectral functions, respectively. The pink dots are experimental photoemission data for SrVO$_3$ [52]. The Fermi level is set at zero energy.

Fig. 7A shows that for $U_V = 5$ eV and $U'_V = 3.5$ eV, Sr$_2$VO$_4$ is metallic with mainly V $t_{2g}$ states at the Fermi level, which is a reproduction of Fig 3B. Fig. 7B shows that with $U_V$ increased to 8 eV and $U'_V$ to 6.8 eV which approximately fixes the $p$-$d$ separation, a metal-insulator transition occurs and Sr$_2$VO$_4$ is rendered a Mott insulator. However, the critical $U_V$ depends on the approximation scheme we employ. A more elaborate cluster-DMFT calculation and/or the inclusion of long range order may find a smaller critical $U_V$ [63].

Appendix B: Mott gap of Sr$_2$MnO$_4$ and its effects on Sr$_2$MnO$_4$/Sr$_2$VO$_4$ superlattices

In this appendix, we show how the Hubbard $U_{Mn}$ changes the Mott gap of Sr$_2$MnO$_4$ with the $p$-$d$ separation approximately fixed. Fig. 8A shows the orbitally resolved spectral function of Sr$_2$MnO$_4$ with $U_{Mn} = 8$ eV and $U'_{Mn} = 7.5$ eV. Note that since Sr$_2$MnO$_4$ is a Mott insulator, the Fermi level in the calculation is shifted at the conduction band edge, i.e. the edge of Mn $e_g$ states. In Fig. 8D of the main text, the Mott gap of Sr$_2$MnO$_4$ is around 0.5 eV with $U_{Mn} = 5$ eV and $U'_{Mn} = 4.5$ eV. If we increase $U_{Mn}$ to 8 eV and $U'_{Mn}$ to 7.5 eV, the Mott gap is correspondingly increased to around 1 eV with the $p$-$d$ separation fixed by the photoemission data [52]. The Mn $t_{2g}$ peak and the main peak of O $p$ states now merge together. However, even with $U_{Mn} = 8$ eV, the Mott gap is still smaller than the optical gap.
(around 2 eV) from experiment [27]. The difference could be due to spatial correlations not included in our single-site DMFT approximation [63].

Using the parameters $U_{\text{Mn}} = 8$ eV and $U'_{\text{Mn}} = 7.5$ eV, we redo the calculations on Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattices (with $U_V = 5$ eV and $U'_V = 3.5$ eV) to test the effects of Mott gap size on charge transfer. As Fig. 8 shows, the key features in electronic structure remain the same as Fig. 4 in the main text: i) Mn $e_g$ and V $t_{2g}$ states emerge at the Fermi level and ii) the main peaks of O p states associated with the MnO$_2$ and VO$_2$ layers are approximately aligned. This shows that it is the p-d separation that controls the charge transfer across the interface while the size of Mott gap plays a secondary role.

Appendix C: Electronic structure calculated using the fully localized limit double counting

In this appendix, we show the electronic structure of both SrVO$_3$/SrMnO$_3$ and Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattices, calculated using the standard fully localized limit (FLL) double counting. The orbitally resolved spectral function is shown in Fig. 9 which is compared to Fig. 4 in the main text. We employ $U_V = U_{\text{Mn}} = 5$ eV. We find the FLL double counting does not change the key features of electronic structure, such as the emergence of Mn $e_g$ and V $t_{2g}$ states at the Fermi level.

Appendix D: (SrVO$_3$)$_1$/(SrMnO$_3$)$_1$ versus (SrVO$_3$)$_2$/(SrMnO$_3$)$_2$ superlattices

In this appendix, we compare the (SrVO$_3$)$_1$/(SrMnO$_3$)$_1$ superlattice to the (SrVO$_3$)$_2$/(SrMnO$_3$)$_2$ superlattice. We focus on the d occupancy and the charge transfer from V to Mn sites. Table III shows that $N_d$ of V sites and Mn sites are very similar between (SrVO$_3$)$_1$/(SrMnO$_3$)$_1$ and (SrVO$_3$)$_2$/(SrMnO$_3$)$_2$ superlattices.
FIG. 8: A) Orbitally resolved spectral function of Sr$_2$MnO$_4$. $U'$ double counting is employed with $U_{Mn} = 8$ eV, $U'_{Mn} = 7.5$ eV. The red (thin), blue (very thick) and green curves (thick) are Mn $t_{2g}$, Mn $e_g$ and O $p$ projected spectral functions, respectively. The pink dots are experimental photoemission data of SrMnO$_3$ [52]. The Fermi level is set at zero energy. B) Orbitally resolved spectral function of Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattices. $U'$ double counting is employed with $U_{Mn} = 8$ eV, $U'_{Mn} = 7.5$ eV and $U_{V} = 5$ eV, $U'_{V} = 3.5$ eV. B1): Mn $t_{2g}$ (red thin) and Mn $e_g$ (blue thick) states; B2): V $t_{2g}$ (green thick) and V $e_g$ (violet thin) states; B3): O $p$ states of the MnO$_2$ layer (turquoise thick) and O $p$ states of the VO$_2$ layer (maroon thin).
FIG. 9: Orbitally resolved spectral function of vanadate-manganite superlattices. Left panels: SrVO$_3$/SrMnO$_3$ superlattice. Right panels: Sr$_2$VO$_4$/Sr$_2$MnO$_4$ superlattice. A) and B): Mn $t_{2g}$ (red thin) and Mn $e_g$ (blue thick) states; C) and D): V $t_{2g}$ (green thick) and V $e_g$ (violet thin) states; E) and F): O $p$ states of the MnO$_2$ layer (turquoise thick) and O $p$ states of the VO$_2$ layer (maroon thin). Fully localized limit double counting is employed with $U_V = U_{Mn} = 5$ eV. The Fermi level is set at zero energy.
TABLE III: The occupancy of V $d$ and Mn $d$ states, as well as VO$_2$ and MnO$_2$ layers in vanadates, manganites and the superlattices. All the occupancies without the parentheses are calculated from Wannier basis using the DFT-LDA relaxed structures. The occupancies in the parentheses are calculated from Wannier basis using the experimental structures.

|                | SrVO$_3$ | SrMnO$_3$ | (SrVO$_3$)$_1$/SrMnO$_3$ | (SrVO$_3$)$_2}$/SrMnO$_3$ |
|----------------|----------|-----------|--------------------------|---------------------------|
| $N_d$(V)       | 2.09     | 0.0       | 1.73                     | 1.69                      |
|                | (2.03)   | (4.08 4.06) | (4.51)                  | (4.54)                    |
| $N$(VO$_2$)    | 13.36    | 4.06      | 12.86                    | 12.82                     |
|                | (13.34)  | (15.35)   | (15.92)                  | (15.95)                   |

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