Using *ab initio* thermodynamics we compile a phase diagram for the surface of Fe₃O₄(001) as a function of temperature and oxygen pressures. A hitherto ignored polar termination with octahedral iron and oxygen forming a wave-like structure along the [110]-direction is identified as the lowest energy configuration over a broad range of oxygen gas-phase conditions. This novel geometry is confirmed in a x-ray diffraction analysis. The stabilization of the Fe₃O₄(001)-surface goes together with drastic changes in the electronic and magnetic properties, e.g., a half-metal-to-metal transition.

PACS numbers: 68.35.Md, 68.35.Bs, 68.47.Gh, 73.20.At, 75.70.Rf, 61.10.Nz

The surface composition and structure of a multicomponent material is a fundamental property that has significant consequences for its surface reactivity, mechanical, and magnetic properties. Magnetite is not only important in geophysics and mineralogy, but is also attracting increasing attention as a potential material for spintronic devices [1, 2], due to its half-metallic behavior, coupled with a high Curie temperature of 858 K.

Magnetite crystallizes in the inverse spinel structure, where the oxygen atoms form a slightly distorted face centered cubic lattice and the iron atoms occupy tetrahedral and octahedral interstitial sites. The stacking sequence in the [001]-direction consists of A-layers containing tetrahedral iron (Fe_A) and B-layers with oxygen and octahedral iron (Fe_B) atoms (cf. Fig. 1).

Two alternative approaches are widely used to assess possible terminations for metal oxide surfaces, an ionic scheme considering the electrostatic energy [3] and the autocompensation rule applied to covalent bonds [4]. After the scheme of Tasker [3], bulk truncations of the Fe₃O₄(001)-surface should be unstable because of a diverging surface energy due to uncompensated dipole moments. Such structures were, therefore, so far discarded. Both the ionic [3] and the covalent [4] approach rely on the information from the bulk only and suggest that stabilization of a polar surface can only be achieved through massive changes of the surface stoichiometry.

For Fe₃O₄(001) a \((\sqrt{2} \times \sqrt{2})\sqrt{45}\)-reconstruction has been observed experimentally [5]. However, the surface structure and stoichiometry are still under debate. Applying the autocompensation rule [4] to Fe₃O₄(001), two models for the \((\sqrt{2} \times \sqrt{2})\sqrt{45}\)-reconstruction have been proposed so far: an A-layer where half of the tetrahedral iron is missing and a B-layer with oxygen vacancies or hydroxyl groups. Several variations of the former model have been suggested: The x-ray photoelectron diffraction (XPD) analysis of Chambers et al. [5] and the low energy ion scattering (LEIS) results of Mijiritskii et al. [6] were interpreted as a half-filled A-layer with strong inward relaxations of the surface layers. Concerning the magnitude of the interlayer distances, there is a considerable discrepancy. In a molecular dynamics calculation with classical potentials - the only theoretical work so far - Rustad et al. [7] suggested a 0.5ML A-termination where the surface and half of the subsurface Fe₃ relax in octahedral positions in the surface B-layer. Also non-autocompensated terminations with a charge ordered B-layer were reported from scanning tunneling microscopy (STM)-measurements using a ferro- [8] or an antiferromagnetic tip [9]. However, the observed magnetic contrast alone cannot explain the LEED- and XRD-superstructure intensities and does not provide a clue on the stabilization mechanism of the surface.

In this Letter we present the results of a systematic investigation of the composition, structure and properties of the Fe₃O₄(001)-surface based on density-functional theory (DFT) calculations. Using *ab initio* atomistic thermodynamics [10, 11] we predict that a "polar" termination is the lowest energy configuration over the entire range of accessible oxygen pressures. The stabilization of the surface involves a fundamentally different mechanism, which has not been considered so far: While most of the previous studies proposed an ordering of surface vacancies as the origin of the surface reconstruction, here it is explained as a Jahn-Teller distortion...
of the surface atoms forming a wave-like pattern along the [110]-direction. A x-ray diffraction (XRD)-analysis supports this theoretically predicted model.

We use the full-potential augmented plane waves (FP-APW) method in the W|EN2K-implementation [15] and the generalized gradient approximation (GGA) [16]. The Fe₃O₄(001)-surface is modeled by a symmetric slab containing five B- and four to six A-layers depending on the structural model. The vacuum between the repeated slabs amounts to 10 Å. We have ensured convergence with respect to the thickness of the slab by repeating the calculations for the most stable configurations with a thicker slab with an additional A- and B-layer on both sides (approximately 100 atoms in the unit cell as opposed to 70 in the original unit cell). The surface free energies for both supercells are within ±3meV/Å² equal. The lateral parameter of the supercell is set to the GGA bulk lattice constant, 8.42 Å, which is in good agreement with the experimental value of 8.394 Å.

The convergence parameters for the mixed APW+lo and LAPW basis set are: \( R_{\text{Fe}}^{\text{MT}} = 1.90 \) bohr, \( R_{\text{O}}^{\text{MT}} = 1.40 \) bohr, inside the muffin tins (MTs) wave functions expansion in spherical harmonics up to \( r_{\text{MT}}^{-10} \) and non-spherical contributions to the electron density and potential up to \( r_{\text{MT}}^{-4} \). The energy cutoff for the plane wave representation in the interstitial is \( E_{\text{wf}}^{\text{max}} = 19 \) Ry for the wave functions and \( E_{\text{wf}}^{\text{max}} = 196 \) Ry for the potential. With these cutoff parameters a convergence of energy differences better than 1 mRy is achieved. Results for the \((\sqrt{2} \times \sqrt{2})R45°\)-unit cell are obtained with 4 \( k_{\parallel} \)-points in the irreducible part of the Brillouin zone.

The configurations we have considered include (i) a 0.5 ML of tetrahedral iron above the B-layer, (ii) or relaxed in the B-layer; (iii) a modified 0.5 ML A-termination where the surface and half of the subsurface FeA occupies octahedral sites in the B-layer [11]. Additionally, we have considered two geometries with one oxygen vacancy per unit cell: either as a nearest neighbor to a tetrahedral iron or above an octahedral iron. Together with the non-autocompensated A- and B-terminations shown in Fig.[4] they were used as a starting point for the DFT-calculations, where we performed a full structural optimization of the atomic positions in the outer two AB-double-layers with damped Newton dynamics [13].

The calculated surface phase diagram for all studied terminations is compiled in Fig. 2. The oxygen chemical potential is converted into oxygen pressures for a fixed temperature of \( T = 900 \) K, which corresponds to a typical annealing temperature in experiment. The range of allowed oxygen chemical potentials lies between the “oxygen-poor”-limit, which corresponds to the case where the oxide would decompose in iron crystallites and oxygen gas and the “oxygen-rich”-limit, defined by the chemical potential of oxygen in the oxygen molecule, set as zero reference. In Fig. 2 the surface energy is given with respect to bulk magnetite. We note that at

\( T = 900 \) K and \( p_{O2} > 10^{-6} \) mbar hematite becomes the thermodynamically stable phase [10]. Taking into account the calculated enthalpy of formation for magnetite (11.32 eV) and hematite (8.024 eV) [20], the stability range of Fe₃O₄ extends up to \( \mu_{O2} = -1.5 \) eV.

Auto-compensated terminations, e.g., the ones containing 0.5 ML FeA in the surface layer, are independent of the oxygen pressure, while the effect of the oxygen environment is most distinct for the two bulk truncations with an A- or a B-layer. In line with electrostatic considerations, the A-termination is unstable and this trend is substantially enhanced for oxygen-rich conditions. A most striking result is that the other bulk truncated surface, the B-termination, which has been so far discarded as a non-compensated surface, turns out to be the most stable configuration over a broad range of oxygen pressures even in the ideal (unrelaxed) geometry. For oxygen-
like the unreconstructed ideal B-termination entails a B-layer. An asymmetric force on the surface oxygens in termination, respectively.

Due to the lateral relaxation of oxygen the Fe distance is reduced from 2.06 Å to 1.96 Å on the surface and are thus closer to the bulk values for the tetrahedral (d_{Fe-O_{bulk}} = 1.89 Å) coordination. Perpendicular to the surface, all surface atoms relax inwards except for O1(S), which relaxes outwards. The atoms in the subsurface A- and B-layer relax outwards. A detailed structural analysis will be presented elsewhere.

A top and side view of the novel surface geometry, is given at the top of Fig. 2. The wave-like-pattern along the [110]-direction emerges from a strong lateral relaxation of 0.30 Å of one of the two oxygen atoms with a FeA-neighbor (marked as O1(S) in Fig. 2) towards the FeB-row and a subsequent alternating shift of FeB-pairs by ±0.09 Å perpendicular to the FeB-row. The oxygen atoms with a FeA-neighbor (O2(S)) relax by 0.14 Å towards and by 0.03 Å parallel to the FeB-row. Due to the lateral relaxation of oxygen the FeB-oxygen distance is reduced from 2.06 Å in the bulk to 1.96 Å on the surface and are thus closer to the bulk values for the tetrahedral (d_{Fe-O_{bulk}} = 1.89 Å) coordination. Perpendicular to the surface, all surface atoms relax inwards except for O1(S), which relaxes outwards. The atoms in the subsurface A- and B-layer relax outwards. A detailed structural analysis will be presented elsewhere.

X-ray measurements were performed on a natural crystal at the beam line DW12 at LURE/Orsay with a wave length of 0.826 Å. In total, 220 symmetrically independent reflections were measured at room temperature in 5 crystal truncation rods (CTR) and 6 superstructure rods. The cleanliness of the sample and the reproducibility of the (2√2 X 2√2)R45°-LEED pattern were verified by a repeated sputtering and annealing up to 850 K in UHV and oxygen pressures up to 5×10⁻⁶ mbar. Furthermore, the LEED I/V curves of the natural and an artificial crystal are nearly identical.

To minimize the effect of surface inhomogeneities only the superstructure reflections were considered in the analysis. The experimental data and the calculated curves for the two most promising terminations, i.e., the half-filled A-layer of tetrahedral iron and the modified B-termination are displayed in Fig. 3. The corresponding R-factors obtained after a structural refinement with the DFT-geometries are 0.24 and 0.19, respectively, which indicates a preference for the modified B-termination. Both models vary only by the additional tetrahedral iron in the half-filled A-layer. The relaxations of the remaining atoms are rather similar leading to an overall similarity in the intensity distribution along q_z. Still the fit with the modified B-layer is noticeably better. The B-termination with oxygen vacancies, proposed by Stanka et al. is incompatible with the XRD-data (R-factor worse than 0.6). The wave-like structure along the [110]-direction observed in the STM-measurements of Stanka et al. is a further evidence for the modified B-termination.

The stabilization of the FeO_4(001)-surface and the unusual lattice distortions are connected with strong changes in the electronic and magnetic properties. In the bulk, magnetite shows a half-metallic behavior with a band gap in the majority spin channel of approximately 0.5 eV and a 100% spin-polarization due to the t_{2g}-states of FeB at E_F in the minority spin channel.

On the surface (cf. Fig. 3) surface states in the majority spin gap reduce the spin-polarization at the Fermi level to approximately 60%. Spin-resolved photoemission measurements support this result. Since electrons near the Fermi level are important for transport properties, we examined closer their nature: the states at E_F are confined to the surface layer and are due to the hybridization between d_{x^2-y^2}-states of FeB and p_x, p_y-states of O1(S). Additionally, there is a noticeable contribution from the tetrahedral iron in the subsurface layer. In order to get more insight into the driving mechanisms towards the reconstruction of the surface we have plotted in Fig. 3 also the difference between the DOS of the B-termination with modified and ideal positions of the atoms. For the modified B-termination a band gap opens in the majority spin channel between −0.4 and −0.2 eV where states of d_{x^2-y^2} (d_{z^2}) character are shifted to higher (lower) energies. The latter cross the Fermi level and eventually become partially unoccupied. Vice versa, in the minority spin channel, bands become occupied. Altogether this
FIG. 4: Total density of states (DOS) of Fe$_3$O$_4$(0001) terminated by a modified B-layer for the majority (minority) spin channel in the upper (lower) panel is shown in grey. The difference of the DOS of the modified B-termination and the bulk-terminated B-layer is given with a black line. A Gauss broadening of $2\sigma = 0.05$ eV is used.

reduces the energy of the system.

This combination of a Jahn-Teller-distortion and a spin-flip leads to interesting magnetic properties. In the bulk magnetite is a ferrimagnet with the magnetic moments of tetrahedral ($-3.43\mu_B$) and octahedral iron (3.50 $\mu_B$) oriented antiparallel to each other and a total magnetic moment of 4.0 $\mu_B$ per formula unit. In the modified B-termination the magnetic moments of the subsurface Fe$_A$ ($-3.45\mu_B$) is close to the bulk value. However, the magnetic moments of the surface Fe$_B$ (3.00 $\mu_B$) is substantially reduced due to the strong relaxations.

Additionally, a substantial magnetic moment of approximately $0.25\mu_B$ is induced in the undercoordinated surface oxygen with a missing bond to tetrahedral iron, while the magnetic moments of the rest of the surface oxygen atoms are close to the bulk value, 0.07 $\mu_B$.

A similar magnetization was observed for the oxygen-terminated $\alpha$-Fe$_2$O$_3$(0001)-surface [20].

Based on ab initio thermodynamics we identify a B-termination with octahedral iron and oxygen in the surface layer forming a wave-like structure along the [110]-direction as the thermodynamically stable configuration over a broad range of oxygen pressures. Instead of ordering of surface defects the symmetry reduction is achieved by a Jahn-Teller distortion and is connected with strong changes of the electronic and magnetic properties, e.g. a metalization at the surface. Experimental evidence for the DFT-predicted geometry is given in STM-measurements [21, 22] and a XRD-analysis performed with the atomic positions obtained with DFT.

Theoretically, “polar” surfaces have been predicted for a number of metal oxide surfaces, e.g. Fe$_2$O$_3$(0001) [20], RuO$_2$(110) [13], ZnO(0001) [24], and PbO [27]. The modified B-termination of Fe$_3$O$_4$(0001) is a further example that excluding such surfaces from the structural analysis can be misleading.

This work is supported by the DFG, PE 883.

* Electronic address: pentcheva@lrz.uni-muenchen.de
† permanent address: MPI für Mikrostrukturphysik, Weinberg 2, 06120 Halle, Germany

[1] M.A.M. Gijs and P.J. Kelly, Europen Patent No. EP 0 672 303 A1 (1995).
[2] W. Eerenstein et al., Phys. Rev. Lett. 88, 247204, (2002).
[3] P.W. Tasker, J. Phys. C 12, 4977 (1979).
[4] J.P. LaFemina, Crit. Rev. Surf. Chem. 3, 297 (1994)
[5] J.M. Gaines et al., Surf. Sci. 373, 85 (1997).
[6] G. Tarrach et al., Surf. Sci. 285, 1 (1993).
[7] S.A. Chambers, S. Thevuthasan and S.A. Joyce, Surf. Sci. 450, L273 (2000).
[8] A.V. Mijiritskii and D.O. Boerma, Surf. Sci. 486, 73 (2001).
[9] B. Stanka et al., Surf. Sci. 448, 49 (2000).
[10] F.C. Voogt et al., Phys. Rev. B 60, 11193 (1999).
[11] J.R. Rustad, E. Wasserman, and A.R. Felmy, Surf. Sci. 432, L583 (1999),
[12] R. Wiesendanger et al., Science 255, 583 (1992).
[13] G. Mariotto, S. Murphy, and I.V. Shvets, Phys. Rev. B 66, 245426 (2002).
[14] C.M. Weinert and M.Scheffler, In: Defects in Semiconductors, H.J. Bardeleben (Ed.), Mat Sci. Forum 10-12,25 (1986).
[15] K. Reuter and M. Scheffler, Phys. Rev. B 65, 035406 (2002).
[16] P. Blaha, K. Schwarz, G.K.H. Madsen, D. Kvasnicka and J. Luitz, WIEN2k, An Augmented Plane Wave + Local Orbitals Program for Calculating Crystal Properties, (Karlsruhein Schwarz, Techn. Univ. Wien, Austria),2001. ISBN 3-9501031-1-2
[17] J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. 77, 3865, (1996).
[18] B. Kohler et al., Comp. Phys. Commun. 94, 31 (1996).
[19] F.D. Richardson and J.H.E. Jeffes, Journal of Iron and Steel Industry 160, 261 (1948).
[20] X.-G. Wang et al., Phys. Rev. Lett. 81, 1038-1041 (1998).
[21] H.A. Jahn and E. Teller, Proc. Roy. Soc. A 161, 220 (1937).
[22] The total energy as a function of the distortion shows a cusp at the symmetric initial configuration and is thus in accord with the Jahn-Teller theorem.
[23] M. Fonin et al., private communication.
[24] R. Pentcheva et al., to be published.
[25] Besides 0.02% Mn no contaminations of the crystal were detected with x-ray fluorescence spectroscopy.
[26] S. Frank, D. Schrupp and W. Moritz, private communication.
[27] STM simulations performed in the Tersoff-Hamann model are consistent with the STM images of Stanka et al. [4]. These results will be published elsewhere.
[28] Z. Zhang and S. Satpathy, Phys. Rev. B 44, 13319 (1991).
[29] O. Dulub, U. Diebold, and G. Kresse, Phys. Rev. Lett. 90, 016102 (2003).
[30] J. Rogal, K. Reuter and M. Scheffler, Phys. Rev. B 69, 075421 (2004).