Supporting Information

Ultrathin structures derived from interfacially modified polymeric nanocomposites to curb electromagnetic pollution

Kumari Sushmita a, Petr Formanek b, Dieter Fischer b, Petra Pötschke b, Giridhar Madras c, Suryasarathi Bose d*

a Centre for Nanoscience and Engineering, Indian Institute of Science, Bangalore- 560012, India

b Leibniz-Institut für Polymerforschung Dresden e. V., Dresden-01069, Germany

c Interdisciplinary Centre for Energy Research, Indian Institute of Science, Bangalore- 560012, India

d Department of Materials Engineering, Indian Institute of Science, Bangalore- 560012, India

*Corresponding author: sbose@iisc.ac.in

1) The conceptual basis for the choice of polymer and solvent

For a preliminary test of miscibility, neat films and multilayered films were cast using a doctor blade setup and observed using digital imaging, as shown in figure S1. Before casting, the individual polymeric solutions/dispersions were prepared. The solvent used for PVDF was dimethylformamide (DMF), and PMMA and PC were chloroform (CHCl₃). As already mentioned, PMMA is used as the interfacial layer for better stitching of the other two layers. PMMA is soluble in both CHCl₃ and DMF, and hence diffusion of the PMMA layer into both the phases of the multilayered assembly can be expected.

A couple of trial experiments were performed by dissolving polymers in various solvents such as dimethylformamide (DMF), tetrahydrofuran (THF), chloroform (CHCl₃), and N-Methyl-2-pyrrolidone (NMP). The combinations of polymer-solvents that were attempted are shown in table S1a. The ratio of solvent to polymer was decided manually based on the viscosity of the resulting solutions, which have to be optimum for the film preparation. The ones with the tick mark in table S1a denote the systems that were practically (visually) found to dissolve at least 1g of polymer in 10 ml solvent (temperature was kept 40 °C for low boiling point solvents such as
CHCl₃ and THF and 100 °C for high boiling point solvents such as DMF and NMP). The multilayered films cast using these solutions were visually inspected, and only a few of the combinations were found to be mechanically stable. These combinations include PVDF-DMF/PMMA-DMF/PC-CHCl₃, PVDF-DMF/PMMA-CHCl₃/PC-CHCl₃, PVDF-DMF/PMMA-THF/PC-THF. The idea was to choose a solvent such that PMMA can diffuse towards both the PVDF and PC sides for better stitching. The final selection of the pairs, PVDF-DMF, PMMA-CHCl₃, and PC-CHCl₃, served the purpose. The multilayered film obtained using these pairs is shown in figure S1f.

Thus, finally, the PVDF solution was prepared by dissolving 2 g PVDF in 10 ml DMF, the PMMA solution was prepared by dissolving 3 g PMMA in 10 ml CHCl₃, and the PC solution was prepared by dissolving 2.5 g PC in 10 ml CHCl₃. The films of individual layers of PC, PMMA, and PVDF were cast using a 300 μm doctor blade, and they were sufficiently dried before immersing it in cold water to peel them off from the glass substrate. The films were further dried in a vacuum oven to evaporate the water, and the final films are shown in figures S1a, S1b, and S1c. Similarly, PVDF/PMMA and PMMA/PC films were sequentially cast using 200 μm and 300 μm blades, and the final films are shown in figures S1d and S1e, respectively. Figure S1f shows the final film obtained by the sequential casting of the PVDF/PMMA/PC layer using 100 μm, 200 μm, and 300 μm blade (slit size). Figure S1g and S1h show the sequential stacking of PVDF followed by PC (or PVDF/PC) and PC followed by PVDF (or PC/PVDF), respectively. Doctor blades with 200 μm and 300 μm slit sizes were used to sequentially stacking these layers. The PMMA layer was purposefully not added in these two cases (figure S1g and S1h) to justify its importance as a mid-layer.

As expected, PC and PMMA films were transparent, as seen in the digital image. PVDF was opaque in appearance. As seen in the images S1d, S1e, and S1f, the films are also transparent. However, figures S1g and S1h showing both combinations of PC and PVDF indicate opaque films. Thus, the miscibility of PMMA with both PC and PVDF has played its role in enhancing the transparency of the multilayered assembly of PVDF/PMMA/PC. The films obtained finally for PVDF/PMMA, PVDF/PC, and PC/PVDF were too brittle for mechanical analysis.
Figure S1: Digital images of (a) PC film, (b) PMMA film, (c) PVDF film, (d) PVDF/PMMA film, (e) PMMA/PC film, (f) PVDF/PMMA/PC film, (g) PVDF/PC film and (h) PC/PVDF film obtained using the doctor blade method.

Table S1b and c show the predicted polymer-solvent interactions based on the Hansen solubility parameters. Table S1b shows the value of Hansen’s solubility parameters obtained from literature\(^1\), whereas table S1c indicates if a particular solvent will dissolve the polymer or not based on equations 1 and 2.
\[ R_a^2 = 4(\delta_{d2} - \delta_{d1})^2 + (\delta_{p2} - \delta_{p1})^2 + (\delta_{h2} - \delta_{h1})^2 \]  \hspace{1cm} (1)

Relative energy difference or RED = \( \frac{R_a}{R_0} \) \hspace{1cm} (2)

where \( \delta_d \) refers to the energy from dispersion forces between molecules, \( \delta_p \) refers to the energy from the dipolar intermolecular force between molecules, \( \delta_h \) refers to the energy from hydrogen bonds between molecules, subscript 1 and 2 stands for solvent and solute and \( R_0 \) refers to the interaction parameter of the polymer.

If

\[ \text{RED} < 1 \rightarrow \text{polymer will dissolve} \]
\[ \text{RED} = 1 \rightarrow \text{polymer will partially dissolve} \]
\[ \text{RED} > 1 \rightarrow \text{polymer will not dissolve} \]

The theoretical prediction holds for most cases except the PVDF-NMP system, which was practically found to be soluble (1g PVDF was soluble in 10 ml NMP), as is also evident from existing literature\(^2\). Moreover, the PC-DMF system was practically poorly/sparingly soluble for concentrations as low as 1 g PC in 10 ml DMF.

Table S1

a) Solubility results of the selected polymers in various solvents by visual inspection

|       | DMF | THF | CHCl\(_3\) | NMP |
|-------|-----|-----|------------|-----|
| PVDF  | ✔   | ✖  | ✖          | ✔   |
| PMMA  | ✔   | ✔  | ✔          | ✔   |
| PC    | ✖ (sparingly soluble) | ✔ (soluble but as solute content increases, solubility drastically decreases) | ✔ | ✔ |

b) Hansen solubility parameter (HSP) and its parts for the studied polymers and solvents \(^1\)

Note that units used are MPa\(^{1/2}\) for the HSP.

|       | Dispersion (\(\delta_d\)) | Polar (\(\delta_p\)) | Hydrogen Bonding (\(\delta_h\)) | Interaction Radius (\(R_0\)) |
|-------|---------------------------|---------------------|-------------------------------|-----------------------------|
| PVDF  | 17.00                     | 12.10               | 10.20                         | 4.10                        |
|        | RED  | Conclusion       |
|--------|------|------------------|
| PVDF-DMF | 0.51 | Soluble          |
| PVDF-THF | 1.65 | Not Soluble      |
| PVDF-CHCl₃ | 2.48 | Not Soluble      |
| PVDF-NMP | 2.39 | Not Soluble      |
| PMMA-DMF | 0.64 | Soluble          |
| PMMA-THF | 0.71 | Soluble          |
| PMMA-CHCl₃ | 0.91 | Soluble          |
| PMMA-NMP | 0.98 | Soluble / partially soluble |
| PC-DMF   | 0.63 | Soluble          |
| PC-THF   | 0.62 | Soluble          |
| PC-CHCl₃ | 0.68 | Soluble          |
| PC-NMP   | 0.77 | Soluble          |

* Note- Hansen solubility parameters may not hold for all systems and might have its limitation.

2) **Characterization of polymers**

This section includes the characterization of neat polymer for reference. Figure S2 shows the Fourier transform infrared (FTIR) spectra of commercial PVDF, PMMA and PC. The peaks corresponding to signature functional groups are highlighted in the respective polymers' FTIR spectra, based on the existing literature ³⁻⁹. The observed spectra confirm that the materials under test are neat PVDF, PMMA and PC. The XRD of commercial PVDF used in this work can be obtained from the existing article published by Bose et al. ¹⁰. The DSC thermogram of commercial PVDF and PC used in this work is already reported by Bose et al. ⁹, ¹⁰. DSC was performed for PMMA and the glass transition temperature was observed to be ~100 °C.
3) Characterization of hybrid nanostructures

Figures S3a and S3b show the low and high magnification SEM micrographs of MoS$_2$-Fe$_3$O$_4$, whereas Figures S3c and S3d show those of rGO-Fe$_3$O$_4$. 
Figures S3: High and low magnification SEM micrographs of (a,b) MoS$_2$-Fe$_3$O$_4$ and (c,d) rGO-Fe$_3$O$_4$.

Figures S4a and S4b show the low magnification TEM micrograph of MoS$_2$-Fe$_3$O$_4$ and rGO-Fe$_3$O$_4$, whereas Figures S4c and S4d show the SAED pattern of both materials. The diffraction rings in the SAED pattern of rGO-Fe$_3$O$_4$ (figure S4d) conforms to the XRD pattern. However, the SAED pattern of MoS$_2$-Fe$_3$O$_4$ (figure S4c) comprises amorphous rings, which are probably because of lower crystallinity, which is in accordance with the XRD pattern where the peaks intensity is comparatively weaker than that of rGO-Fe$_3$O$_4$. 
Figure S4: (a) Low magnification TEM micrographs of MoS$_2$-Fe$_3$O$_4$, (b) Low magnification TEM micrographs of rGO-Fe$_3$O$_4$, (c) SAED pattern of MoS$_2$-Fe$_3$O$_4$ and (d) SAED pattern of rGO-Fe$_3$O$_4$.

Figure S5 shows the digital images of PVDF (rGO-Fe$_3$O$_4$), PVDF (MoS$_2$-Fe$_3$O$_4$), PC(CNT), PVDF(rGO-Fe$_3$O$_4$)/M/PC, and PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC films obtained using the doctor blade method. It is to be noted that PVDF+ 10 wt% rGO-Fe$_3$O$_4$ obtained using doctor blade approach is designated as PVDF (rGO-Fe$_3$O$_4$), PVDF+ 10 wt% MoS$_2$-Fe$_3$O$_4$ is designated as PVDF (MoS$_2$-Fe$_3$O$_4$), PC+ 3 wt% CNT is designated as PC(CNT), the multilayered assembly with rGO-Fe$_3$O$_4$ is designated as PVDF(rGO-Fe$_3$O$_4$)/M/PC, and that with MoS$_2$-Fe$_3$O$_4$ as PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC. The thin films were obtained by casting and subsequent procedures, as shown in scheme 3. These digital images show that PVDF film, after incorporating 10 wt% nanofiller, resulted in a film with poor film quality. In contrast, the multilayered films obtained using this technique showed superior homogeneity and mechanical property. Though PC(CNT)
does not show a visible wear, it was structurally inferior for mechanical testing because of PC composite's brittle nature at such low thickness.

![Digital images of films](image)

Figure S5: Digital images of (a) PVDF (rGO-Fe$_3$O$_4$) film, (b) PVDF (MoS$_2$-Fe$_3$O$_4$) film, (c) PC (CNT) film, (d) PVDF(rGO-Fe$_3$O$_4$)/M/PC film, (e) PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC film obtained using the doctor blade method.

Figure S6 shows the EDS mapping of PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC. The sample was prepared with a rough approach. It was cut using a razor blade, and the cross-section was analyzed. The main idea was to understand the circular light grey regions in the SEM micrograph. From the EDS mapping shown in figure S6d, it is evident that those light grey regions are PVDF. It is also noted that these light grey circular regions are more prominent in samples cut by a razor blade than in samples prepared by ultramicrotome. Hence, we chose the approach using a razor blade for the analysis here. These light grey regions can be the crystallites of PVDF or are manifested during the fracture process towards the side of PVDF. Nevertheless, the area of this light grey matter is PVDF.
Figure S6: a) SEM micrograph of razor blade cut sample of PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC b) EDS mapping of carbon c) EDS mapping of oxygen d) EDS mapping of fluorine e) EDS mapping of Sulphur.

4) Additional dielectric and EM shielding results

Figure S7 shows the real permittivity of PC(CNT), PVDF(rGO-Fe$_3$O$_4$)/M/PC, and PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC films. The given samples real permittivity is in the frequency range of $10^{-1}$- $10^{7}$ is positive. By lowering the frequency, the real permittivity rises. The found result of PC(CNT) is quite different from the result reported before by our group $^{11}$, where the real permittivity showed negative values at low frequency. This indicates that the real permittivity is dependent not only
on the thickness of the sample but also on the concentration and state of dispersion of the nanoparticles in the PC matrix.

Figure S7: Real permittivity versus frequency.

Figures S8 shows SE$_T$ as a function of frequency for one stack of PVDF+ 10 wt% rGO-$\text{Fe}_3\text{O}_4$ \{or [PVDF(rGO-$\text{Fe}_3\text{O}_4$)]$^1$\} and PVDF+ 10 wt% MoS$_2$-$\text{Fe}_3\text{O}_4$ \{or [PVDF(MoS$_2$-$\text{Fe}_3\text{O}_4$)]$^1$\}. It is to be noted that the superscript in the nomenclature denotes the number of stacks (here it is 1). SE$_T$ is found to be -0.85 dB for [PVDF(rGO-$\text{Fe}_3\text{O}_4$)]$^1$ and -0.01 dB for [PVDF(MoS$_2$-$\text{Fe}_3\text{O}_4$)]$^1$ at 26.5 GHz.
Figure S8: $SE_T$ versus frequency for [PVDF(rGO-Fe$_3$O$_4$)]$_1$ and [PVDF(MoS$_2$-Fe$_3$O$_4$)]$_1$ in X band, Ku band, and K band.

Figures S9 illustrates the percentage of absorption/ reflection for [PC(CNT)]$_1$, [PVDF(rGO-Fe$_3$O$_4$)/M/PC]$_1$, and [PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC]$_1$ at 8.2 and 26.5 GHz frequency. The percentage of absorption for [PC(CNT)]$_1$ is found to be 71.9, [PVDF(rGO-Fe$_3$O$_4$)/M/PC]$_1$ is found to be 81.9, and [PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC]$_1$ is found to be 76.4 at 26.5 GHz frequency. The percentage absorption for [PC(CNT)]$_1$ is higher compared to [PC(CNT)]$_9$. As CNTs amount is 9-fold when using [PC(CNT)]$_9$, reflection-based shielding overpowered the absorption-based shielding. However, this is not the case for the multilayered assemblies of PVDF(rGO-Fe$_3$O$_4$)/M/PC and PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC, where absorption-based shielding dominates even on increasing the number of layers.
Figure S9: $SE_A$ and $SE_R$ for [PC(CNT)]$^1$, [PVDF(rGO-Fe$_3$O$_4$)/M/PC]$^1$ and [PVDF(MoS$_2$-Fe$_3$O$_4$)/M/PC]$^1$ at 8.2 and 26.5 GHz frequency.

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