Skyrmions in twisted van der Waals magnets

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Magnetic skyrmions in 2D chiral magnets are in general stabilized by a combination of Dzyaloshinskii-Moriya interaction and external magnetic field. Here, we show that skyrmions can also be stabilized in twisted moiré superlattices in the absence of an external magnetic field. Our setup consists of a 2D ferromagnetic layer twisted on top of an antiferromagnetic substrate. The coupling between the ferromagnetic layer and the substrate generates an effective alternating exchange field. We find a large region of skyrmion crystal phase when the length scales of the moiré periodicity and skyrmions are compatible. Unlike chiral magnets under magnetic field, skyrmions in moiré superlattices show enhanced stability for the easy-axis (Ising) anisotropy which can be essential to realize skyrmions since most van der Waals magnets possess easy-axis anisotropy.

I. INTRODUCTION

The discovery of ferromagnetism in two-dimensional (2D) monolayer CrI\textsubscript{3} and other 2D van der Waals (vdW) materials opened a new window for exploring low dimensional magnetism and its applications in spintronics\textsuperscript{[10]} The properties of 2D materials can be controlled by external parameters\textsuperscript{[10][12]} and are highly sensitive to stacking and twisting between the layers\textsuperscript{[10][15][16]}. In particular, with the discovery of superconductivity in twisted bilayer graphene\textsuperscript{[21][23]} there has been tremendous progress on exploring moiré superlattices both experimentally and theoretically\textsuperscript{[19][24]}. In terms of magnetism, stacking order and twisting can significantly alter the interlayer exchange as the exchange is highly sensitive to atomic registries\textsuperscript{[25][30]}.

Magnetic skyrmions\textsuperscript{[31]} are nanoscale vortex-like spin textures that were first observed in non-cetrosymmetric bulk magnetic materials such as MnSi\textsuperscript{32,33}, (FeCo)Si\textsuperscript{34} and FeGe\textsuperscript{35}. Skyrmions are topologically protected, hence they can not be deformed continuously into other magnetic states. In recent years skyrmions received ample attention due to their potential for spintronics applications and memory storage device\textsuperscript{[36]}. In most cases, skyrmions are stabilized by interplay of Dzyaloshinskii-Moriya (DM) interaction and external magnetic field\textsuperscript{[37][38][39]}. In this article, we explore the possibility of stabilizing magnetic skyrmions in the absence of an external magnetic field in moiré superlattices. We consider a FM monolayer on an AFM substrate. Twisting the FM layer by an angle \( \theta \) produces moiré patterns as shown in Fig. 1(a). (c). Ferromagnetic coupling between the substrate and the FM monolayer leads to an alternating exchange field for the moiré superlattice as shown in Fig. 1(b). Our setup is motivated from Ref. \textsuperscript{[30]} However unlike Ref. \textsuperscript{[30]} which includes dipole-dipole interaction to stabilize magnetic skyrmions, we consider DM interaction which is the primary interaction for magnetic skyrmions in chiral magnets\textsuperscript{[40][41]}. Our main results are summarized in Fig. 2 and 3. We show that (i) skyrmion crystal (SkX) is stabilized as a function of exchange coupling between the layers \((J_{ex})\) and moiré periodicity. (ii) Even though SkX can be stabilized for a wide range of twisting angle, we find the optimal moiré periodicity to be about \( L = 9L_{D} \), where \( L_{D} = (J/D)\alpha \) is the intrinsic length scale for skyrmions. (iii) Unlike chiral magnets under magnetic field, we find an extended region of SkX for easy-axis anisotropy. (iv) We show that the topological charge \( q = \frac{1}{4\pi} \int d^{2}r \hat{m} \cdot \left( \partial_{\alpha} \hat{m} \times \partial_{\alpha} \hat{m} \right) \) of the magnetic skyrmions are concentrated at the edges and splits into three parts for large moiré periodicity and large easy axis anisotropy. This effect arises due to the anisotropic shape of the skyrmion.

II. MODEL

Before we delve into the analysis of the effective magnetic Hamiltonian, we first describe our setup. As mentioned above, we follow the procedure first described by Ref. \textsuperscript{[30]} which considers a FM monolayer twisted on top of an AFM substrate. For twisting angle \( \theta \), the moiré period is given by \( L = a/2\sin(\theta/2) \) where \( a \) is the lattice constant. For small angle \( \theta \), \( L \approx a/\theta \) (large period), the local atomic registries on length scale smaller than \( L \) but larger than \( a \) matches the...
FIG. 2: $A = 0$ phase diagram at different moiré periods and exchange field. (a) Moiré period $L$ vs maximum interlayer exchange field $B_{\text{Max}}$ phase diagram with spiral (Sp) and skyrmion crystal (SkX) phases. Here $L_D = (J/D)a$, where $a$ is the lattice constant. (b,c,d) Magnetization texture and (e,f,g) topological charge density in $1 \times 2$ moiré super cells for $L/L_D = (5, 8, 11)$ and $B_{\text{Max}} J/D^2 = 1.73$ as marked by $\star$, $\blacktriangle$, $\blacksquare$ symbols in (a).

atomic stacking of different interlayer translation $\mathbf{r}$ as shown in Fig.[1][c]. Hence moiré superlattice can be described by interlayer translation vector $\mathbf{r}(\mathbf{R})$ that gives atomic registry at position $\mathbf{R}$. The interlayer exchange coupling between AFM substrate and FM layer is different at different positions due to different atomic stacking of monolayer and substrate and this leads to spatially dependent exchange field $\mathbf{B}(\mathbf{R})$ as shown in Fig.[1][b]. For example, at position $\mathbf{R}_1$ the coupling aligns the spins of 2D layer (green) in positive $z$-direction when it sits on top of AFM sublattice with spins up (black) and the spins align in opposite direction at $\mathbf{R}_2$ when it is on top of AFM sublattice with spins down (red). The interlayer exchange field at interlayer translation $\mathbf{r}$ is given by

$$\mathbf{B}(\mathbf{r}) = \sum_{j,\tau} J_{\mathbf{r},\tau}^{j} (\mathbf{r} - \mathbf{R}_j) \mathbf{m}_{j,\tau}.$$  

(1)

where $J_{\mathbf{r},\tau}$ is the interlayer coupling coefficient, $\mathbf{m}$ is the magnetic moment of top most layer of AFM substrate, $(\tau, \tau') = \{A, B\}$ represents the two inequivalent sites in unit cell, $\xi_A = 0, \xi_B = \{0, a\}$ and the summation $j$ is over the Bravais lattice. The total interlayer exchange field per unit cell is given by summing the fields of sites $A$ and $B$

$$\mathbf{B}(\mathbf{r}) = \mathbf{B}^A(\mathbf{r}) + \mathbf{B}^B(\mathbf{r}).$$  

(2)

This approximation holds when the interlayer coupling is small as compared to intralayer coupling. We used the following coupling form that decays exponentially at long distances

$$J_{\mathbf{r},\tau}^{j} (\mathbf{r} - \mathbf{R}_j) = J_{0}^{j} e^{-\sqrt{\mathbf{r}^2 + d^2}/\xi_0}$$  

(3)

where $d$ is the interlayer separation and $\xi_0$ is the decay length. In our calculations we used $d = a$ and $\xi_0 = 1$.

Next we describe our model for the monolayer. We consider a magnetic model for 2D honeycomb lattice which is relevant to 2D vdW magnets such as trihalides

$$H = -J \sum_{r,\mu} \mathbf{S}_r (\mathbf{S}_{r+\delta_\mu} - D \sum_{r,\mu} [\hat{d}_\mu (\mathbf{S}_r \times \mathbf{S}_{r+\delta_\mu})])$$

$$-A_c \sum_{r,\mu} [(\mathbf{S}_r \cdot \hat{d}_\mu) (\mathbf{S}_{r+\delta_\mu} \cdot \hat{d}_\mu)] + A_s \sum_r (\mathbf{S}_r^z)^2$$

$$- \sum_r \mathbf{B}(\mathbf{R}_r) \cdot \mathbf{S}_r$$  

(4)

where $\mathbf{S}_r^z$ is the local moment at site $r$ and $\hat{d}_\mu$ is the three nearest neighbors on the honeycomb lattice. $J$ is the ferromagnetic Heisenberg exchange coupling, $D$ is the DM coupling. $A_c$ and $A_s$ are the compass and single-ion anisotropies respectively. The DM vector $\hat{d}_\mu = \hat{z} \times \hat{d}_\mu$ is set by the symmetry and originates due to the inversion symmetry breaking on the surface. $\mathbf{B}(\mathbf{R})$ is the interlayer exchange field with the twisted substrate. To explore the phase diagram of $H$, we consider the free energy functional in the continuum $\mathcal{F}[\mathbf{m}] = \int d^2 \mathbf{r} F(\mathbf{m})$ where $\mathbf{m}(\mathbf{r})$ is the local magnetization. The free energy density has the following four components

$$F(\mathbf{m}) = F_{\text{iso}} + F_{\text{DM}} + F_{\text{aniso}} + F_{\text{moire}}$$  

(5)

where

$$F_{\text{iso}} = F_0(\mathbf{m}) + \frac{3}{2} (J/2) \sum_\alpha (\nabla \mathbf{m}^\alpha)^2$$  

(6)
We absorb factor $\frac{3}{2}$ in $J$, $D$ and $A_c$ and define the effective anisotropy $A = A_c + A_s$ which can be positive (easy plane) or negative (easy axis or Ising). To obtain the ground state spin configuration $\mathbf{m}$, we solve the coupled Landau-Lifshitz-Gilbert (LLG) equations\[35\]

$$F_{DM} = \frac{3}{2} D (m^x \partial_x m^x - m^x \partial_x m^y)
+ \frac{3}{2} D (m^y \partial_y m^z - m^y \partial_y m^x)$$

$$F_{aniso} = -\frac{3}{2} A_c [(m^x)^2 + (m^y)^2] + A_s (m^z)^2 + \frac{3A_c}{4} [m^y (\partial_x m^y + \partial_y m^x)
- m^x (\partial_x m^x - \partial_y m^y)]$$

$$F_{moire} = -\mathbf{B}(\mathbf{R}) \cdot \mathbf{m}.$$  

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$$\frac{d\mathbf{m}}{dt} = -\gamma \mathbf{m} \times \mathbf{B}^{eff} + \alpha \mathbf{m} \times \frac{d\mathbf{m}}{dt},$$

where $\mathbf{B}^{eff} = -\delta H/\delta \mathbf{m}$, $\gamma$ is gyromagnetic ratio and $\alpha$ is Gilbert damping coefficient. We start from different initial states and compare the energies of final states to get the actual ground state. To solve LLG equations numerically we used mid point method\[32\] by discretizing the effective magnetic field $\mathbf{B}^{eff} = -\delta H/\delta \mathbf{m}$ on a $1 \times 2$ moiré supercells. We used $L_D = J/D = 10a$ to construct the phase diagrams and the results were also verified at various points for larger values of $L_D$. The magnitude of the magnetization was kept constant at each grid point after each time step enforcing the hard spin constraint, $|\mathbf{m}|^2 = 1$ and periodic boundary conditions were imposed at the boundaries.

**III. RESULTS**

We start by exploring the interplay between the moiré periodicity $L$ and $B_{Max}$ as shown in Fig. [2] for $A = 0$. $B_{Max}$ is the maximum value of interlayer exchange magnetic field. At low exchange field, we get a spiral phase. As we increase the field, the SkX phase starts at the moiré period $L/L_D \approx 9$ which corresponds to the optimum angle between the layer and the substrate. As we further increase the field, we get SkX for a range of values around optimum period. This range increases with increasing the exchange field. Unlike skyrmions in chiral magnets where the size of the skyrmion is set by $L_D$, here we find that the size of skyrmions is determined by the moiré period. On the other hand, $L_D$ determines the boundary length between the interior and the exterior of the skyrmions. For small moiré period, skyrmions are small and their shape is nearly circular as shown in Fig. [2](b). Fig. [2](c) shows that as the period increases, the size of skyrmion also increases and it takes the triangular shape of the exchange field. The corners of skyrmions get sharper with increasing $L$. Unlike the skyrmions in chiral magnets, we find that a large fraction of the topological charge is concentrated at the corners of the skyrmion. This fraction increases with increasing $L$. There is also a small fraction of opposite charge between the skyrmions which decreases with increasing $L$. This charge arises due to the anti-vortices between the skyrmion.\[30\] For large $L$, the topological charge further splits into three parts due to the triangular shape of the skyrmion as shown in Fig. [2](g).

Next, we explore the effects of anisotropy, $A$ at around optimal angle $L = 8L_D$ as shown in Fig. [3]. At low exchange field, we obtain spiral phase for a wide range of $A$, a small

FIG. 3: $L = 8L_D$ phase diagram. (a) Anisotropy $A$ versus maximum interlayer exchange field $B_{Max}$ phase diagram with ferromagnetic (FM), spiral (Sp), skyrmion crystal (SkX) and mixed state (Sp+FM). Here $L_D = (J/D)a$, where $a$ is the lattice constant. (b,c,d) Magnetization texture in $1 \times 2$ moiré supercells and (e,f,g) spin structure factor for ($\star$, $\triangle$, $\square$) symbols marked in (a). The parameters corresponding to these symbols are the following: $\star = \{A = -0.6D^2/J, B_{Max} = 1.15D^2/J\}$, $\triangle = \{A = 0.3D^2/J, B_{Max} = 0.98D^2/J\}$ and $\square = \{A = 1.14D^2/J, B_{Max} = 0.58D^2/J\}$. 

\[7\]
FIG. 4: Evolution of magnetization texture, topological charge density \( \chi \) and spin structure factor \( I(Q) \) for \( AJ/D^2 = \{ -1.2, -0.6, 0 \} \) at \( L = 8L_D \).

The ferromagnetic phase at lowest negative values of \( A \) as well as a mixed (FM+Sp) phase at largest positive values of \( A \). As the field increases, initially we get SkX near the lowest negative values of \( A \) that corresponds to easy axis anisotropy. By further increasing the field, the range of SkX gradually increases and eventually, occupies the whole phase diagram. Fig. 3(b,c,d) show the local magnetization and Fig. 3(e,f,g) show the spin structure factor \( I(Q) \propto |\langle m_Q \rangle|^2 \) for the three phases. \( m_Q \) is the Fourier transform of the magnetization. Unlike an isotropic SkX which has six peaks on the circle in the spin structure factor, we find four peaks lie on circle and two peaks lie inside the circle. This is due to the anisotropic triangular shape of the SkX. The spiral has two peaks at \( Q = \pm Q_0 \) and the mixed state (FM+Sp) has three peaks including the \( Q = 0 \) from the FM and \( Q = \pm Q_0 \) from the spiral phase. The intensity of \( Q = 0 \) peak increases with increasing \( A \) and decreases with increasing exchange field \( B_{Max} \). On the other hand, the intensities corresponding to the spiral wave vectors have the opposite behavior of \( Q = 0 \) with \( A \) and \( B_{Max} \).

The properties of SkX also depends on the anisotropy \( A \). Fig. 4 shows the magnetization, spin structure factor and topological charge density as a function of \( A \) for moiré period \( L = 8L_D \). For \( A = \pm 1.2D^2/J \) the skyrmion has a sharp boundary wall where magnetization changes abruptly and then it changes slowly inside the skyrmion. The sharpness of boundary wall decreases with increasing \( A \) and the change in magnetization inside the skyrmion increases with increasing \( A \). For \( A = -1.2D^2/J \), a large fraction of topological charge is concentrated at the boundary wall of skyrmion and a small fraction lies inside the skyrmion. There is also a small fraction of opposite charge between the skyrmions. The concentration of charge at boundary decreases with increasing \( A \) and the central charge increases with increasing \( A \). The fraction of opposite charge between the skyrmions also increases with increasing \( A \).

We also studied the effects of anisotropy for a non-optimal angle at \( L = 5L_D \). As shown in Fig. 5, SkX is highly suppressed in this case but still persists for large exchange field and easy axis anisotropy. Suppression of SkX is due to the fact that the moiré supercell is too small with respect to the optimal size of the skyrmions.

FIG. 5: Anisotropy \( A \) vs maximum interlayer exchange field \( B_{Max} \) phase diagram with ferromagnetic (FM), spiral (Sp), skyrmion crystal (SkX) and mixed state (Sp+FM) phases at \( L = 5L_D \). Here \( L_D = (J/D)a \), where \( a \) is the lattice constant.

IV. CONCLUSION

We have shown that skyrmion crystal can be stabilized in moiré superlattices in the absence of external magnetic field. We found a large SkX phase for easy axis anisotropy which can be essential to stabilize skyrmions in vdW magnets such as CrI$_3$. In particular, for optimal moiré periodicity SkX occupies majority of the phase diagram. We find that the properties of the skyrmion can be tuned with the moiré periodicity and anisotropy. Unlike skyrmions in chiral magnets, the topological charge is concentrated at the edges of the skyrmion.

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