Switching Characteristics and High-Temperature Dielectric Relaxation Behaviours of Pb(Zn$_{1/3}$Nb$_{2/3}$)$_{0.91}$Ti$_{0.09}$O$_3$ Single Crystal

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Abstract: This work evaluated the resistance switching characteristics in the (100)-oriented Pb(Zn$_{1/3}$Nb$_{2/3}$)$_{0.91}$Ti$_{0.09}$O$_3$ (PZNT) single crystal. The current hysteresis can be closely related to the ferroelectric polarization and we provided a possible explanation using a model about oxygen vacancies to analyze the mechanism of switching. The obvious frequency dispersion of the relative permittivity signified the relaxer-type behavior of the sample. The value of the relaxation parameter $\gamma = 1.48$ was estimated from the linear fit of the modified Curie-Weiss law, indicating the relaxer nature. High-temperature dielectric relaxation behaviors were revealed in the temperature region of 400–650 $^{\circ}$C. In addition, under the measuring frequency of 10 kHz, $\varepsilon_r$ was tunable by changing the electric field and the largest tunability of $\varepsilon_r$ reached 14.78%. At room temperature, the high pyroelectric coefficient and detectivity figure of merit were reported.

Keywords: PZNT single crystal; switching characteristics; oxygen vacancies; dielectric relaxation; pyroelectric coefficient

1. Introduction

With flash memory approaching its scaling limit, the research and development of memory are urgent. Recently, various materials and novel devices have been studied for overcoming the scaling limit of conventional silicon-based memories [1]. New memory concepts include, for instance, magnetic random access memory (MRAM), ferroelectric random access memory and resistance switching random access memory (RRAM). A RRAM device is basically a two-terminal resistor device and it is attractive for its high speed, improved endurance, retention and excellent scalability for use in low-power nonvolatile memory applications. The resistance switching occurs between the low-resistance state and high-resistance state due to the electrically altered change in the resistance [2,3]. Some oxides and complex oxides, such as TiO$_2$–$x$, ZnO$_2$, NiO, TaO$_2$, SrTiO$_3$, Pt$_{0.7}$Ca$_{0.3}$MnO$_3$, etc. [4–15], which have resistance switching characteristics, are under intense investigation due to their potential application in next-generation non-volatile memory [2,16]. In addition, the application of multiferroic materials has also gained great attention in resistance switching [17,18]. BiFeO$_3$ is the best representative [19–24]. In the past two decades, unpoled relaxer ferroelectric (1 – $x$)Pb(Mg$_{1/3}$Nb$_{2/3}$)$_3$O$_3$–$x$PbTiO$_3$ and (1 – $x$)Pb(Zn$_{1/3}$Nb$_{2/3}$)$_3$O$_3$–$x$PbTiO$_3$ solid solutions have been extensively studied because of their excellent dielectric properties, piezoelectric properties and electromechanical properties compared with state-of-the-art lead zirconate titanate ceramics [25]. According to Bertram and Reck [26], at room temperature, (1 – $x$)Pb(Zn$_{1/3}$Nb$_{2/3}$)$_3$O$_3$–$x$PbTiO$_3$ exists in the rhombohedral (FE–) R3m phase when the value of $x$ is less than 0.06. The rhombohedral
R3m phase coexisted with the tetragonal (FE) P4mm phase when the value of x was between 0.08 and 0.11. In addition, when x was greater than 0.11, (1 − x)Pb(Zn_{1/3}Nb_{2/3})O_{3}−xPbTiO_{3} presented a purely tetragonal P4mm phase. Otherwise, a metastable monoclinic phase (Pm) formed during the cooling process after crystal growth in the composition range of x between 0.06 and 0.11. Park and Shrout reported that (1 − x)Pb(Zn_{1/3}Nb_{2/3})O_{3}−xPbTiO_{3} single crystals were near the morphotrophic phase boundary (MPB) when the value of x was about 0.09 [27]. They pointed out that Pb(Zn_{1/3}Nb_{2/3})_{0.91}Ti_{0.09}O_{3} (PZNT) single crystal exhibited two main phases: a rhombohedral-tetragonal ferroelectric (FE) phase between room temperature and 180 °C, and a cubic paraelectric (PE) phase above 180 °C. The PZNT single crystal had two major structures near the MPB: the rhombohedral phase (R3m) and the tetragonal phase (P4mm), and the temperature of the rhombohedral-tetragonal phase transition was about 90 °C. The phase transition could be induced by varying the temperature, electric field or hydrostatic pressure [25,28,29]. Bao et al. reported the dielectric relaxation in the PZNT single crystal at low temperatures, and the relaxation process could be attributed to the freezing of ferroelectric macrodomain walls induced by the pinning of point defects [30].

Although many works regarding the PZNT single crystal have already been published, the resistance switching characteristics and high-temperature dielectric relaxation phenomenon for the PZNT single crystal were beyond the interest of researchers up to now. In this work, the resistance switching characteristics and mechanism were discussed, and the high-temperature dielectric relaxation phenomenon, the dielectric tunability and the pyroelectric properties of the (100)-oriented PZNT single crystal were also reported.

2. Experimental Section

The PZNT single crystal was grown using a modified Bridgman method [27]. The sample oriented along the (100) direction was prepared with dimensions of 5 mm × 5 mm × 1 mm and electroded with silver.

The polarization-electric-field (P–E) hysteresis loops was obtained using a Sawyer-Tower circuit by applying a sinusoidal input signal with a frequency of 1 Hz. The reference capacitor used in the measurement is 5 µF. The I–V curve was measured by a Radiant Technologies Precision premier II Ferroelectric Tester (Albuquerque, NM, USA). The temperature dependence of relative permittivity was measured by a precision LRC meter (Agilent E4980A, Agilent Technologies Inc., Santa Clara, CA, USA) in the temperature range of 20–650 °C with the heating rate of 2 °C/min. The biased temperature dependence of dielectric response was measured using a blocking circuit, a dc power source (Keithley 6517A, Keithley Instruments Inc., Solon, OH, USA) and a multi–frequency LCR meter (Model SR720 of Stanford Research Systems, Stanford University, Stanford, CA, USA) at frequencies of 10 kHz and linear temperature change was adopted as 1 °C/min in the biased heating from the temperature range of 25–240 °C. The tunability was measured using a blocking circuit, a dc power source (Keithley 6517A) and a multi-frequency LCR meter (Model SR720 of Stanford Research System) at room temperature.

Furthermore, the temperature dependence of pyroelectric coefficient was measured by a dynamic technique [32,33]. At a certain temperature T_0, the sample temperature was sinusoidally modulated T(t) = T_0 + T_0 \sin 2\pi ft ] with frequency f = 5 mHz and amplitude T_0 = 1 K using a Peltier element. The pyroelectric current signal was amplified with an electrometer and the 90° out of phase component of current with respect to the temperature modulation was measured with a lock-in amplifier. After setting to a new temperature T_0 the sample was kept at T_0 for 15 min for the signal to become stable before the pyroelectric measurement was performed. PZNT single crystal was first poled at room
temperature by applying an electric field of amplitude 10 kV/cm for 5 min and then short circuited circuited at room temperature overnight.

3. Results and Discussion

The current-voltage (I–V) curves for the (100)-oriented PZNT single crystal obtained by decreasing the sweep range step by step are shown in Figure 1a. In the range of ±50 V, the test voltage was swept at a constant rate from 0–50 V, then to −50 V, before returning to 0 V. The numbers in the figure denote the sequence of the voltage sweeps. An asymmetry in the current can be observed in the figure. In addition, we can find that the I–V curves show distinct hysteresis behavior, indicating the sample exhibited obvious resistance switching characteristics. Moreover, in the inset (A) of Figure 1a, the I–V segment shows an obvious diode-like rectifying I–V characteristic, indicating a diode behavior.

The I–V curves plotted on semilogarithmic scales are shown in the inset (B) of Figure 1a. When the sweeping positive bias increased from 0 V to about 33.3 V, the resistance switching was in the high-resistance state (HRS), with a low current. When the voltage was beyond 33.3 V, the current increased quickly and attained a maximum (around 5 µA) with the voltage reached 50 V, indicating that the resistance switching “turns on” and switches from the HRS to the low-resistance state (LRS). As the voltage swept back from 50 V to about −33.3 V, the resistance switching remained in the LRS. As the resistance switching continued its negative bias, the current stabilized at around 0.7 µA until the sweeping bias exceeded −33.3 V, which led to a transition from the LRS to HRS. Finally, when the sweeping voltage returned, going from −50 to 0 V, the resistance switching remained in the HRS [34,35]. In addition, the resistance switching ratio was 229 at 2.08 V.

The mechanism of resistance switching characteristics is not yet perfectly understood. Despite this, there were some phenomenological models to explain these characteristics, namely the Schottky barrier model [36,37], space charge [38,39], the electrically conducting filamentary model [40,41], and the Mott transition [42]. Jeon et al. [43] reported that resistance switching characteristics are caused by the change in the Schottky potential based on the results from the first-principles calculations. It is worth mentioning that the observed ferroelectric resistance switching behavior in our crystal would be different from that observed in some ferroelectric tunneling junctions [44,45], because the tunneling current can only be taken into account for the ultrathin ferroelectric materials, not for our crystal with a thickness of 1 mm.

![Figure 1](image-url)

**Figure 1.** (a) The I–V curves for the PZNT single crystal by decreasing the sweep range step by step; Inset (A) shows the obvious diode-like rectifying I–V characteristic, indicating a diode behavior; Inset (B) shows the I–V curves plotted on semilogarithmic scales; (b) The P–E hysteresis loops for the PZNT single crystal with various voltages.

In the range of ±12, ±15 and ±20 V, the I–V curve was almost symmetrical and without hysteresis, as shown in Figure 1a. Then, as the voltage increased to ±50 V, obvious hysteresis and resistance...
switching characteristics were observed. It was indicated that the current hysteresis and diode-like behavior can be triggered and switched at a high applied field. Figure 1b shows the $P-E$ hysteresis loops for the PZNT single crystal with various voltages, indicating that the sample exhibits good ferroelectricity. It is worth mentioning that the $P-E$ hysteresis loops starts at around ±45 V (not marked in the figure). According to the report of BiFeO$_3$ [46], because the current and ferroelectric polarization exhibited the hysteresis phenomenon at a high applied field, the current hysteresis can be closely related to the ferroelectric polarization.

The polarization mechanism of ferroelectric materials includes displacement polarization and turning-direction polarization. In the process of the preparation of samples, lead vacancies will appear inevitably due to the volatility of lead, leading to the oxygen vacancies (OVs) due to charge neutrality restrictions. It is well known that the ionization of OVs will create conducting electrons in perovskite-structure oxides during the process of preparation at high temperatures, written as:

\[ V_{0} \rightarrow V_{0} + e' \quad (1) \]
\[ V_{0} \rightarrow V_{0aa} + e' \quad (2) \]

where $V_{0}$, $V_{0aa}$ are single-ionized and double-ionized OVs, respectively. The current occurs by electron injection from one electrode affected by the concentration of OVs near the interface. So, according to the current study, we preferred to believe that the resistance switching characteristics of the PZNT single crystal could be caused by the change in the oxygen vacancies concentration at the metal/oxide interface by the electrically controlled electron injection.

In order to describe the processes underpinning resistance switching, the displacement and migration of OVs near the bottom electrode (BE) were introduced. Displacement was defined as the reversible movement of OVs under electrical bias (the OVs cannot get enough energy to go over the Schottky barrier), where upon removal of the applied voltage the OVs return to their initial locations. Migration occurred when the field-driven movement of the vacancies was not reversible upon removal of the bias (the OVs can get enough energy to go over the Schottky barrier) [47].

A model, shown in Figure 2a–d, was set up to explain the resistance switching characteristics of the PZNT single crystal. The yellow circles with plus signs in the model represented OVs. When the sweeping positive bias increased from 0 V to 33.3 V, the OVs could not get enough energy to go over the Schottky barrier. The positive bias displaces OVs towards BE (displacement of OVs) to enhance the electron injection, as shown in Figure 2a. However, when the bias was beyond 33.3 V, the OVs surmounted the Schottky barrier to BE (migration of OVs) because they obtained enough energy, and it greatly increased the electron injection, as shown in Figure 2b. Consequently, the current increased quickly, the resistance switching “turned on” and the switches changed their state from the HRS to the LRS. The state would remain until the bias reversal. As the voltage swept from 0 V to around −33.3 V, the state still remained at the LRS, but the current appeared asymmetric due to the displacement of OVs, as shown in Figure 2c. When the negative bias kept increasing, the resistance switching returned to the HRS because of the migration of the OVs. Finally, the resistance switching returned to the original state when the bias returned to 0 V.

\[ e, e' \]
\[ HRS \rightarrow LRS \]

\[ \text{Displacement} \]
\[ \text{Migration} \]

\[ \text{Schottky barrier} \]

\[ \text{Set} \]

\[ \text{Reset} \]

\[ \text{BE} \]

\[ \text{P–E hysteresis loops starts at around ±45 V (not marked in the figure).} \]

Figure 2. The displacement of OVs (a) and migration of OVs (b) under positive bias to enhance the electron injection; The displacement of OVs (c) and migration of OVs (d) under negative bias to reduce the electron injection.
In the heating temperature range from 20–650 °C at a rate of 2 °C min\(^{-1}\), the temperature dependence of the relative permittivity \(\varepsilon_r\) and the dielectric loss tan\(\delta\) of the PZNT single crystal at various frequencies (from 1–100 kHz) are shown in Figure 3. The inset shows the dielectric loss tan\(\delta\) in the temperature range between 20–300 °C. Permittivity curves for various frequencies exhibit organized (the relative permittivity \(\varepsilon_r\) decreased with the increasing frequency) and the dielectric peaks are located at the temperature of about 177 °C (temperature of the maximum dielectric permittivity \(T_m\)) and \(T_m\) is unchanged with the increasing frequency, indicating a phase transition from the FE phase to the PE cubic phase. Seung-Eek Park et al. reported that the phase transition temperature was 180 °C [25]. In the temperature of \(T_m\), the values of the dielectric peak \(\varepsilon_m\) could be as high as 8420 at the measurement frequency of 1 kHz. With increasing the frequency from 1–100 kHz, the value of \(\varepsilon_m\) decreased from 8420 to 6135. Specially, there was another dielectric anomaly peak that took place at about 90 °C, indicating a phase transition from the FE\(_r\) phase to the FE\(_t\) phase. The same phenomenon had also been reported by Hosono et al. [48]. Obviously, we can observe (inset of Figure 3) the same phase transformation from the dielectric constant. However, it was worth noting that, at the high temperature range (400–650 °C), the permittivity curves presented a high-temperature relaxation phenomenon that looked like the behavior of a diffuse phase transition and the value of tan\(\delta\) became large, because the space charge polarization or the conductivity of the insulating ceramics increased with the increase in temperature.

![Figure 3. The relative permittivity \(\varepsilon_r\) and dielectric loss tan\(\delta\) as a function of temperature for the PZNT single crystal measured at different frequencies. The inset shows the dielectric loss tan\(\delta\) in the temperature range between 20–300 °C.](image)

From Figure 3, the obvious frequency dispersion of the dielectric constant can be observed, and this phenomenon signifies the typical relaxer behavior of the present specimen. The dielectric characteristics of relaxor ferroelectric materials are well known to deviate from the typical Curie-Weiss behavior, and can be described by a modified Curie-Weiss relationship [49]:

\[
1/\varepsilon_r - 1/\varepsilon_m = (T - T_m)^\gamma / C_1, 1 \leq \gamma \leq 2
\]  

(3)

where \(\gamma\) and \(C_1\) are assumed to be constant. For \(\gamma = 1\), a normal Curie-Weiss law was obtained, and a complete diffuse phase transition was described for \(\gamma = 2\) [50]. The plots of ln\((1/\varepsilon_r - 1/\varepsilon_m)\) versus ln\((T - T_m)\) with a frequency of 10 kHz are shown in Figure 4. We can get the value of \(\gamma = 1.48\) by fitting the experimental data. Values of \(\gamma\) in this work were found to vary from 1.31 to 1.62 in the frequency range from 1–300 kHz. This also supported the evidence of the relaxer nature of the PZNT single crystal.
was 0 kV/mm, the maximum relative permittivity ($\varepsilon_m$) of the PZNT single crystal was 6934 at about 190 °C and the $\varepsilon_m$ decreased from 6934 to 2970 with the applied field increasing from 0 to 1 kV/mm. Particularly, the $T_m$ presented a different value under different applied electric fields, suggesting that the single crystal underwent a second-order phase transition. The values of $\varepsilon_m$ and $T_m$ are listed in Table 1. The result clearly indicates a broadening of the dielectric peak due to the diffuse ferroelectric-paraelectric phase transition.

At 100 Hz, the temperature dependence of $\varepsilon_r$ for the PZNT single crystal during the heating process under different electric fields is shown in Figure 5. The curves of the temperature dependence of $\varepsilon_r$ for the PZNT single crystal were lower and wider as the electric field increased, indicating that the single crystal went through a field-induced phase transition. When the value of the applied field was 0 kV/mm, the maximum relative permittivity ($\varepsilon_m$) was 6934 at about 190 °C and the $\varepsilon_m$ decreased from 6934 to 2970 with the applied field increasing from 0 to 1 kV/mm. Particularly, the $T_m$ presented a different value under different applied electric fields, suggesting that the single crystal underwent a second-order phase transition. The values of $\varepsilon_m$ and $T_m$ are listed in Table 1. The result clearly indicates a broadening of the dielectric peak due to the diffuse ferroelectric-paraelectric phase transition.

![Figure 4](image-url)  
**Figure 4.** Plot of ln($1/\varepsilon_r - 1/\varepsilon_m$) as a function of ln($T - T_m$) for the PZNT single crystal measured at 10 kHz.

![Figure 5](image-url)  
**Figure 5.** Temperature dependence of the relative permittivity ($\varepsilon_r$) of the PZNT single crystal under various electric fields at 100 Hz.

| Electric Field (V/mm) | 0    | 600  | 800  | 1000 |
|----------------------|------|------|------|------|
| $\varepsilon_m$      | 6934 | 4779 | 4179 | 2970 |
| $T_m$ (°C)           | 190  | 206  | 214  | 218  |

**Table 1.** The maximum relative permittivity and corresponding temperature under different electric fields.

Dielectric tunable materials have a wide range of applications, such as in phase shifters, oscillators, filters, etc. [51]. From the above discussion, $\varepsilon_r$ was found to be tunable by changing the electric field. The $\varepsilon_r$ and the tunability of $\varepsilon_r$ are shown in Figure 6, respectively. The test was conducted under
different electric fields at 10 kHz at room temperature and the applied electric fields increased from 0 to 1000 V/mm. The tunability of $\varepsilon_r$ is defined as $(\varepsilon_r(0) - \varepsilon_r(E)) \times 100% / \varepsilon_r(0)$, where $\varepsilon_r(0)$ and $\varepsilon_r(E)$ are the $\varepsilon_r$ values when the electric field is zero and $E$, respectively. The results revealed that the largest tunability of $\varepsilon_r$ was 14.78%. With the increase of the electric field, the $\varepsilon_r$ decreased gradually and the tunability increased, respectively [52,53].

Figure 6. At room temperature, 10 kHz of AC bias and composition dependence of (a) the relative permittivity $\varepsilon_r$; (b) the tunability of $\varepsilon_r$ of the PZNT single crystal.

The pyroelectric coefficients as a function of the temperature for the PZNT single crystal are shown in Figure 7. The real part of the pyroelectric coefficients increases slowly from 447.4 to 576.1 C/m²K with increasing temperatures from 18–50 °C, respectively. The imaginary part of the pyroelectric coefficients are pyroelectric losses. At a room temperature of 25 °C, the value of the pyroelectric coefficient was $-463.3 \mu$C/m²K, and the absolute value of the pyroelectric coefficient (463.3 μC/m²K) was higher than that of the graded PZT films on Pt-coated silicon substrates (202–250 μC/m²K) and Ba₃Nb₂O₈ ceramic (103 μC/m²K) [32,54]. A useful comparative figure of merit (FOM) used in comparing pyroelectric materials is defined as $D_F = |P| / (c\varepsilon_r(0)\tan\delta)^{1/2}$, where $c$ is the heat capacity per unit volume (c = 2.5 J/cm³K) [55], $\varepsilon_r$ is the relative permittivity, $\varepsilon_0$ is the permittivity of the vacuum, and $\tan\delta$ is the dissipation factor. The $D_F$ value is $8.77 \times 10^{-6}$ Pa⁻⁰.⁵ at 1 kHz. This result implies that the PZNT single crystal can be a promising material for pyroelectric array sensor applications.

Figure 7. The pyroelectric coefficient as a function of temperatures for the PZNT single crystal.
4. Conclusions

In conclusion, this work has shown the resistance switching characteristics in the (100)-oriented PZNT single crystal. The current hysteresis can be closely related to the ferroelectric polarization and we provided a possible explanation with a model of oxygen vacancies to analyze the mechanism of switching. In the process of heating, the temperature of the rhombohedral-tetragonal phase transition was about 90 °C and the temperature of the FE-PE phase transition was about 177 °C. The obvious frequency dispersion of the relative permittivity signified the relaxer-type behavior of the sample, and the value of the relaxation parameter $\gamma = 1.48$, estimated from the linear fit of the modified Curie-Weiss law, indicated the relaxer nature. High-temperature dielectric relaxation behaviors were revealed in the temperature region of 400–650 °C. In addition, under the measuring frequency of 10 kHz, we found that $\varepsilon_r$ was tunable by changing the electric field and the largest tunability of $\varepsilon_r$ reached 14.78%. At room temperature, the pyroelectric coefficient and the figure of merit $F_D$ were $-463.3 \, \mu C/m^2 K$ and $8.77 \times 10^{-6} \, Pa^{-0.5}$, respectively.

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References
1. Meijer, G.I. Materials science—Who wins the nonvolatile memory race? Science 2008, 319, 1625–1626. [CrossRef] [PubMed]
2. Waser, R.; Aono, M. Nanoionics-based resistive switching memories. Nat. Mater. 2007, 6, 833–840. [CrossRef] [PubMed]
3. Panda, D.; Huang, C.Y.; Tseng, T.Y. Resistive switching characteristics of nickel silicide layer embedded HfO$_2$ film. Appl. Phys. Lett. 2012, 100, 112901. [CrossRef]
4. Ghenzi, N.; Rozenberg, M.J.; Llopis, R.; Levy, P.; Hueso, L.E.; Stoliar, P. Tuning the resistive switching properties of TiO$_2$ films. Appl. Phys. Lett. 2015, 106, 123509. [CrossRef]
5. Santhanakrishna, A.K.; Takshi, A. Photoelectric Memory Effect in an Organic Bulk Heterojunction Device. J. Phys. Chem. C 2015, 119, 17253–17259. [CrossRef]
6. Li, J.C.; Hou, X.Y.; Cao, Q. Effect of Cu doping on the resistive switching of NiO thin films. J. Appl. Phys. 2014, 115, 164507. [CrossRef]
7. Salomone, L.S.; Kasulin, A.; Carbonetto, S.H.; Garcia-Inza, M.A.; Redin, E.G.; Berbeglia, F.; Lipovetzky, J.; Faigón, A.; Campabadal, F. Radiation and bias switch-induced charge dynamics in Al$_2$O$_3$-based metal-oxide-semiconductor structures. J. Appl. Phys. 2014, 116, 174506. [CrossRef]
8. Chen, Y.C.; Chung, Y.L.; Chen, B.T.; Chen, W.C.; Chen, J.S. Revelation on the Interrelated Mechanism of Polarity-Dependent and Multilevel Resistive Switching in TaO$_x$-Based Memory Devices. J. Phys. Chem. C 2013, 117, 5758–5764. [CrossRef]
9. Zhang, Y.; Deng, N.; Wu, H.; Yu, Z.; Zhang, J.; Qian, H. Metallic to hopping conduction transition in Ta$_2$O$_5$-TaO$_y$ resistive switching device. Appl. Phys. Lett. 2014, 105, 063508. [CrossRef]
10. Huang, Z.C.; Chen, Q.; Zhai, Y.; Wang, J.L.; Xu, Y.B. Oxygen vacancy induced magnetization switching in Fe$_2$O$_4$ epitaxial ultrathin films on GaAs(100). Appl. Phys. Lett. 2015, 106, 182401. [CrossRef]
11. Sim, H.J.; Choi, D.; Lee, D.; Seo, S.; Lee, M.-J.; Yoo, I.-K.; Hwang, H. Resistance-switching Characteristics of polycrystalline Nb$_2$O$_5$ for nonvolatile memory application. IEEE Electron. Dev. Lett. 2005, 26, 292–294.
12. Kubicek, M.; Schmitt, R.; Messerschmitt, F.; Jennifer, L.M.R. Uncovering Two Competing Switching Mechanisms for Epitaxial and Ultrathin Strontium Titanate-Based Resistive Switching Bits. ACS Nano 2015, 9, 10737–10748. [CrossRef] [PubMed]
13. Park, J.H.; Kwon, D.H.; Park, H.G.; Jung, C.U.; Kim, M. Role of oxygen vacancies in resistive switching in Pt/Nb-doped SrTiO$_3$. Appl. Phys. Lett. 2014, 105, 183103. [CrossRef]
14. Guo, Z.L.; Sa, B.S.; Zhou, J.; Sun, Z.M. Role of oxygen vacancies in the resistive switching of SrZrO₃ for resistance random access memory. J. Alloys Compd. 2013, 580, 148–151. [CrossRef]

15. Seong, T.G.; Choi, K.B.; Seo, I.T.; Oh, J.H.; Moon, J.W.; Hong, K.; Nam, S. Resistive switching properties of amorphous PtₓC₀.₃MnO₃ films grown on indium tin oxide/glass substrate using pulsed laser deposition method. Appl. Phys. Lett. 2012, 100, 212111. [CrossRef]

16. Sawa, A. Resistive switching in transition metal oxides. Mater. Today 2008, 11, 28–36. [CrossRef]

17. Chen, X.; Tang, X.; Chen, X.-Z.; Chen, Y.-L.; Guo, X.; Ge, H.-X.; Shen, Q.-D. Nonvolatile data storage using mechanical force-induced polarization switching in ferroelectric polymer. Appl. Phys. Lett. 2015, 106, 042903. [CrossRef]

18. Xie, Z.K.; Yue, Z.; Ruehl, G.; Peng, B.; Zhang, J.; Yu, Q.; Zhang, X.; Li, L. Bi(Ni₁/₂Zr₁/₂)O₃-PbTiO₃ relaxor-ferroelectric films for piezoelectric energy harvesting and electrostatic storage. Appl. Phys. Lett. 2014, 104, 245902. [CrossRef]

19. Guo, E.J.; Herklotz, A.; Roth, R.; Christl, M.; Das, S.; Widdra, W.; Dörr, K. Tuning the switching time of BiFeO₃ capacitors by electrodes’ conductivity. Appl. Phys. Lett. 2013, 103, 022905. [CrossRef]

20. Yan, F.; Xing, G.Z.; Li, L. Low temperature dependent ferroelectric resistive switching in epitaxial BiFeO₃ films. Appl. Phys. Lett. 2014, 104, 132904. [CrossRef]

21. Zhao, B.Y.; Chen, Z.; Meng, J.; Lu, H.; Zhang, D.W.; Jiang, A. Ferroelectric polarization and defect-dipole switching in an epitaxial (111) BiFeO₃ thin film. J. Appl. Phys. 2015, 117, 204103. [CrossRef]

22. Zhang, L.X.; Chen, J.; Cao, J.L.; He, D.Y.; Xing, X.R. Large resistive switching and switchable photovoltaic response in ferroelectric doped BiFeO₃ single crystals by chemical solution deposition. J. Mater. Chem. C 2015, 3, 4706–4712. [CrossRef]

23. Shuai, Y.; Ou, X.; Wu, C.; Zhang, W.; Zhou, S.; Bürger, D.; Reuther, H.; Slesazeck, S.; Mikolajick, T.; Helm, M.; et al. Substrate effect on the resistive switching in BiFeO₃ thin films. J. Appl. Phys. 2012, 111, 07D906–07D908. [CrossRef]

24. Park, S.E.; Shroul, T.R. Characteristics of relaxor-based piezoelectric single crystals for ultrasonic transducers. IEEE Trans. Ultrason. Ferroelectr. Freq. Control 1997, 44, 1140–1147. [CrossRef]

25. Bertram, R.; Reck, G.; Uecker, R. Growth and correlation between composition and structure of (1 – x)Pb(Zn₁/₃Nb₂/₃)O₃-xPbTiO₃ crystals near the morphotropic phase boundary. J. Cryst. Growth 2003, 253, 212–220. [CrossRef]

26. Fang, B.J.; Xu, H.Q.; He, T.H.; Luo, H.S.; Yin, Z.W. Growth mechanism and electrical properties of [Pb(Zn₁/₃Nb₂/₃)₀.₉₁Ti₀.₀₉]O₃ single crystals by a modified Bridgman method. J. Crystal. Growth 2002, 244, 318–326. [CrossRef]

27. Zhang, H. Polarization rotation and field induced phase transition in PZN-4.5%PT single crystal. J. Appl. Phys. 2013, 114, 084109. [CrossRef]

28. Zhang, H. Polarization rotation and phase transition in [100]-oriented PZN-PT single crystals determined by the electro-crystalline anisotropy. J. Appl. Phys. 2013, 114, 184102. [CrossRef]

29. Bao, P.; Yan, F.; Dai, Y.; Zhu, J.; Wang, Y.; Luo, H. Dielectric relaxation in 91%Pb(Zn₁/₃Nb₂/₃)O₃–9%PbTiO₃ single crystal at low temperature. Appl. Phys. Lett. 2004, 84, 5317–5319. [CrossRef]

30. Pilch, M.; Molak, A. Resistivity switching induced in ferroelectric phase of PbTiO₃ studied by XPS and electric conductivity tests. J. Alloys Compd. 2014, 586, 488–498. [CrossRef]

31. Tang, X.G.; Jiang, L.L.; Kuang, S.J.; Ding, A.L.; Chen, H.L.W. Dielectric and pyroelectric properties of compositionally graded Pb(Zr₀.₇Ti₀.₃)O₃ thin films prepared by sol-gel process. Chin. J. Chem. Phys. 2007, 20, 665–669. [CrossRef]

32. Tang, X.G.; Han, H.L.W.; Ding, A.L. Electrical properties of compositionally graded lead calcium titanate thin films. Solid State Commun. 2003, 127, 625–628. [CrossRef]

33. Wouters, D.J.; Waser, R.; Wuttig, M. Phase-Change and Redox-Based Resistive Switching Memories. Proc. IEEE 2015, 103, 1274–1288. [CrossRef]

34. Qiu, J.; Shiib, A.; Zhou, W.D.; Mi, Z.T.; Shiib, I. Effects of metal contacts and dopants on the performance of ZnO-based memristive devices. J. Appl. Phys. 2011, 110, 014513. [CrossRef]
36. Fujii, T.; Kawasaki, M.; Sawa, A.; Akoh, H.; Kawazoe, Y.; Tokura, Y. Hysteric Current-Voltage Characteristics and Resistance Switching at an Epitaxial Oxide Schottky Junction SrRuO$_3$/SrTi$_{0.99}$Nb$_{0.01}$O$_3$. *Appl. Phys. Lett.* **2005**, *86*, 012107. [CrossRef]

37. Sawa, A.; Fujii, T.; Tokura, Y. Hysteretic Current-Voltage Characteristics and Resistance Switching at a Rectifying Ti/Pt$_{0.7}$Ca$_{0.3}$MnO$_3$ Interface. *Appl. Phys. Lett.* **2004**, *85*, 4073–4075. [CrossRef]

38. Lee, M.H.; Kim, K.M.; Kim, G.H.; Seok, J.Y.; Song, S.J.; Yoon, J.H.; Hwang, C.S. Study on the Electrical Conduction Mechanism of Bipolar Resistive Switching TiO$_2$ Thin Films Using Impedance Spectroscopy. *Appl. Phys. Lett.* **2010**, *96*, 152909. [CrossRef]

39. Shang, D.S.; Wang, Q.; Chen, L.D.; Dong, R.; Li, X.M.; Zhang, W.Q. Effect of Carrier Trapping on the Switchable Diode Effect and Microwave Composite Ceramics. *Jpn. J. Appl. Phys.* **2004**, *43*, 8663–8666. [CrossRef]

40. Kim, D.C.; Seo, S.; Ahn, S.E.; Suh, D.-S.; Lee, M.J.; Park, B.-H.; Yoo, I.K.; Baek, I.G.; Kim, H.-J.; Yim, E.K.; et al. Electrical Observations of Filamentary Conductions for the Resistive Memory Switching in NiO Films. *Appl. Phys. Lett.* **2006**, *88*, 202102. [CrossRef]

41. Rohde, C.; Choi, B.J.; Jeong, D.S.; Choi, S.; Zhao, J.-S.; Hwang, C.S. Identification of a Determining Parameter for Resistive Switching of TiO$_2$ Thin Films. *Appl. Phys. Lett.* **2005**, *86*, 262907. [CrossRef]

42. Fujimoto, M.; Koyama, H.; Konagai, M.; Hosoi, Y.; Ishihara, K.; Ohnishi, S.; Awaya, N. TiO$_2$ Anatase Nanolayer on TiN Thin Film Exhibiting High-Speed Bipolar Resistive Switching. *Appl. Phys. Lett.* **2006**, *89*, 223509. [CrossRef]

43. Jeon, S.H.; Park, B.H.; Lee, J.; Lee, B.; Han, S. First-principles Modeling of Resistance Switching in Perovskite Oxide Material. *Appl. Phys. Lett.* **2006**, *89*, 042904. [CrossRef]

44. Maksymovych, P.; Jesse, S.; Yu, P.; Ramesh, R.; Baddorf, A.P.; Kalinin, S.V. Polarization Control of Electron Tunneling into Ferroelectric Surfaces. *Science* **2005**, *309*, 1421–1425. [PubMed]

45. Garcia, V.; Fusil, S.; Bouzehouane, K.; Enouz-Vedrenne, S.; Mathur, N.D.; Barthelemy, A.; Bibles, M. Giant tunnel electroresistance for non-destructive readout of ferroelectric states. *Nature* **2009**, *460*, 81–84. [CrossRef]

46. Wang, C.; Jin, K.J.; Xu, Z.T.; Wang, L.; Ge, C.; Lu, H.B.; Guo, H.Z.; He, M.; Yang, G.Z. Switchable diode effect and ferroelectric resistive switching in epitaxial BiFeO$_3$ thick films. *Appl. Phys. Lett.* **2011**, *98*, 192901. [CrossRef]

47. Mojarrad, S.A.; Goss, J.P.; Kwa, K.S.K.; Zhou, Z.Y.; Al-Hamadany, R.A.S.; Appleby, D.J.; Ponon, N.K.; O’Neill, A. Leakage current asymmetry and resistive switching behavior of SrTiO$_3$. *Appl. Phys. Lett.* **2012**, *101*, 173507. [CrossRef]

48. Hosono, Y.; Harada, K.; Kobayashi, T.; Itsumi, K.; Izumi, M.; Yamashita, Y.; Ichinose, N. Temperature dependence of dielectric and piezoelectric properties of Pb(Zn$_{1/3}$Nb$_{2/3}$)$_2$O$_3$-PbTiO$_3$ piezoelectric single crystals. *Jpn. J. Appl. Phys.* **2003**, *42*, 7084–7088. [CrossRef]

49. Zhang, T.F.; Tang, X.G.; Huang, X.X.; Liu, Q.X.; Jiang, Y.P.; Zhou, Q.F. High-Temperature Dielectric Relaxation Behaviors of Relaxer-Like PbZrO$_3$-SrTiO$_3$ Ceramics for Energy-Storage Applications. *Energy Technol.* **2016**, *4*, 633–640. [CrossRef]

50. Zhang, J.J.; Zhai, J.W.; Zhang, Y.W. Microwave and infrared dielectric response in BaTiO$_3$ based relaxor ferroelectrics. *Phys. Status Solidi A* **2011**, *208*, 2853–2860. [CrossRef]

51. Chou, X.J.; Zhai, J.W.; Yao, X. Dielectric tunable properties of low dielectric constant Ba$_{0.5}$Sr$_{0.5}$TiO$_3$-Mg$_2$TiO$_4$ microwave composite ceramics. *Appl. Phys. Lett.* **2007**, *91*, 122903–122908. [CrossRef]

52. Feng, Z.Y.; Zhao, X.Y.; Luo, H.S. Composition and orientation dependence of phase configuration and dielectric constant tunability in poled Pb(Mg$_{1/3}$Nb$_{2/3}$)$_2$O$_5$-PbTiO$_3$ single crystal. *J. Phys. Condens. Matter* **2004**, *16*, 6771–6778. [CrossRef]

53. Tang, X.G.; Chew, K.-H.; Chan, H.L.W. Diffuse phase transition and dielectric tunability of Ba(Zr$_{1-x}$Ti$_x$)$_2$O$_7$ relaxor ferroelectric ceramics. *Acta Mater.* **2004**, *52*, 5177–5183. [CrossRef]

54. Pati, B.; Choudhary, R.N.P.; Das, P.R.; Parida, B.N.; Padhee, R. Pyroelectric and dielectric properties of lead-free ferroelectric Ba$_5$Nb$_2$O$_8$ ceramic. *J. Alloy. Compd.* **2014**, *592*, 6–11. [CrossRef]

55. Whatmore, W.R.; Patel, A.; Sharrock, N.M.; Ainger, F.W. Ferroelectric materials for thermal ir sensors state-of-the-art and perspectives. *Ferroelectrics* **1990**, *104*, 269–283. [CrossRef]