Use of thulium-sensitized rare earth-doped low phonon energy crystalline hosts for IR sources

Joseph Ganem1* and Steven R Bowman2

Abstract
Crystalline hosts with low phonon energies enable novel energy transfer processes when doped with rare earth ions. Two applications of energy transfer for rare earth ions in thulium-sensitized low phonon energy crystals that result in infrared luminescence are discussed. One application is an endothermic, phonon-assisted cross-relaxation process in thulium-doped yttrium chloride that converts lattice phonons to infrared emission, which raises the possibility of a fundamentally new method for achieving solid-state optical cooling. The other application is an optically pumped mid-IR phosphor using thulium-praseodymium-doped potassium lead chloride that converts 805-nm diode light to broadband emission from 4,000 to 5,500 nm. These two applications in chloride crystals are discussed in terms of critical radii calculated from Forster-Dexter energy transfer theory. It is found that the critical radii for electric dipole-dipole interactions in low phonon energy chloride crystals are comparable to those in conventional oxide and fluoride crystals. It is the reduction in multi-phonon relaxation rates in chloride crystals that enable these additional energy transfer processes and infrared luminescence.

Keywords: Rare earth-doped crystals; Thulium cross-relaxation; Solid-state optical cooling; Mid-infrared sources; Optically pumped mid-infrared phosphors; Potassium lead chloride

PACS: 42.72.Ai; 78.60.Lc; 63.20.kd

Background
Rare earth-doped crystals are widely used in many applications that require sources of visible and near-infrared radiation. However, when doped into conventional commercially available crystals such as YAG or YLF, rare earth ions do not radiate efficiently at wavelengths much longer than 3 μm. The mid-infrared range (3 to 10 μm) is not directly accessible using host crystals that have tightly bound oxygen or fluorine ions. The reasons are the relatively high energies for lattice phonons in these crystals and the fact that the rates for non-radiative multi-phonon relaxation increase exponentially as the energies of the electronic transitions are reduced and fewer phonons are required to bridge the gap. The demand for mid-infrared sources and applications in gas detection, remote sensing, IR spectroscopy, and infrared countermeasures has motivated research on alternative methods for generating mid-infrared. Quantum cascade lasers [1], thermal tungsten filaments, small bandgap III-V or II-VI optically pumped semi-conductors [2,3], rare earth-doped chalcogenide glasses [4], oxide glasses [5], and rare earth-doped fluoride crystals [6] have all been used as sources of mid-infrared. This paper discusses an approach to generating mid-infrared that uses rare earth-doped crystals with reduced phonon energies. It focuses specifically on crystals sensitized for diode pumping with the trivalent rare earth ion thulium (Tm3+). This ion is interesting because it has a strong absorption near 800 nm that is readily accessible using low-cost laser diodes, and it has cross-relaxation mechanisms between the pumped state and the ground state that produce two infrared-emitting Tm3+ ions for every ion activated by the pump.

Figure 1 illustrates the lower energy levels of Tm3+ that become populated when the 3H4 level is pumped near 800 nm. Populations of the first three excited states can be monitored by observing infrared fluorescence near 1,400 nm from the 3H4, near 1,200 nm from the 3H5, and near 1,800 nm from the 3F4. Two non-resonant phonon-assisted cross-relaxation mechanisms involving the thulium ion are responsible for this behavior. The first is a non-resonant, phonon-assisted cross-relaxation process that converts thulium lattice phonons to infrared emission, which is the subject of the next section. The second is that the 3H4 state is not directly pumped by 800 nm laser diodes because it lies above the ground state 3F4.

* Correspondence: Ganem@loyola.edu
1 Department of Physics, Loyola University Maryland, 4501 N. Charles Street, Baltimore, MD 21210, USA
Full list of author information is available at the end of the article

© 2013 Ganem and Bowman; licensee Springer. This is an open access article distributed under the terms of the Creative Commons Attribution License (http://creativecommons.org/licenses/by/2.0), which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited.
pumped $^3\text{H}_4$ state and the ground $^3\text{H}_6$ state are also illustrated in Figure 1.

The ‘two for one’ cross-relaxation process labelled C$_1$ that feeds the $^3\text{F}_4$ state is well known for Tm$^{3+}$ and has been used in YAG [7] and YLF [8] host crystals to sensitize 2-$\mu$m sources for diode pumping. Diode-pumped lasers near 2-$\mu$m using singly doped Tm$^{3+}$:YAG and co-doped Tm$^{3+}$:Ho$^{3+}$:YAG are in wide use. However, incorporating Tm$^{3+}$ into a host crystal with reduced multi-phonon relaxation rates enables emission from the $^3\text{H}_5$ state that is fed by the cross-relaxation process labelled C$_2$. In contrast, for conventional Tm$^{3+}$, 2-$\mu$m laser material multi-phonon quenching of the $^3\text{H}_5$ leads to strong heat generation and distortions. Reduced multi-phonon quenching in low phonon energy materials also results in additional energy transfer processes when Tm$^{3+}$ is co-doped with other species of rare earth ions.

This paper discusses two results that arise from Tm$^{3+}$ cross-relaxation in low phonon energy host crystals: (1) In singly doped crystals with Tm$^{3+}$, the C$_2$ process is a phonon-assisted cross-relaxation channel that is endothermic and converts lattice phonons into infrared emission. This raises the possibility of a fundamentally new way of achieving solid-state optical cooling. (2) In crystals co-doped with Tm$^{3+}$ and Pr$^{3+}$, cross-relaxation results in efficient energy transfer to the lower energy levels of the Pr$^{3+}$ ions that fluoresce in the mid-infrared. The result is an optically pumped phosphor that converts 800-nm diode light into mid-IR emission between 4 and 5.5 $\mu$m.

While these two results have different motivations, the underlying physical mechanisms are the same. Both results involve sensitized luminescence using diode-pumped Tm$^{3+}$ ions in host crystals with reduced multi-phonon relaxation rates. The purpose of this paper is to show how crystals with low phonon energies enable these novel energy transfer processes.

**Methods**

**Sensitized luminescence**

In an insulator, excited-state ions can transfer energy non-radiatively to ions of the same species or of a different species through a distance-dependent electric dipole-dipole interaction [9]. The energy transfer rate between sensitizers and acceptors $W_{sa}$ for two ions separated by a distance $R$ can be written in the form

$$W_{sa} = W_s(R_{cr}/R)^6,$$

where $W_s$ is the rate at which the sensitizer ions radiate and $R_{cr}$ the critical interaction distance, is the distance between the sensitizers and acceptors for which the energy transfer rate $W_{sa}$ is equal to $W_s$. In other words, for two ions separated by the critical distance $R_{cr}$, the probability of a sensitizer ion radiating is equal to the probability of its energy transfer to an acceptor ion. Therefore, crystals in which sensitizers and acceptors are on average closer than the critical radius, $W_{sa} > W_s$, which results in non-radiative energy transfer being favoured over radiation.

The critical interaction distance $R_{cr}$ is given by Dexter’s formula [10]:

$$R_{cr}^6 = (3/4\pi)(hc/n^2)Q_s\int f_s^{em}(E) f_a^{abs}(E)dE.$$

In this expression, $n$ is the index of refraction, $Q_s$ is the integrated absorption cross section of the acceptor ion $\sigma(E)dE$, and $f_s^{em}$ and $f_a^{abs}$ are the normalized ($\int f(E)dE = 1$) emission and absorption spectra with $E$ the photon energy equal to $hc/\lambda$. This means that the greater the overlap between the sensitizer ion’s emission spectrum and the acceptor ion’s absorption spectrum, the greater the critical distance. A large critical distance allows a relatively dilute distribution of sensitizer and acceptor ions within the lattice to interact and exchange energy at rates faster than their radiative rates.
The practical consequence of Dexter’s formula is that the energy transfer is much more likely in a system in which there is significant overlap between the excited-state transitions of the sensitizing ions and the ground-state absorptions of the acceptor ions. Even in a singly doped system, in which the acceptors and sensitizers are of the same species, the pump will only interact with a small fraction of the total ions available. This means that the average distance between an excited-state ion and a ground-state ion is essentially equal to the average distance $R_{av}$ between the ions in the crystal, assuming a random distribution is given by

$$R_{av} = (1/N)^{1/3},$$

(3)

where $N$ is the density of ions in the lattice. If $R_{av}$ is less than or equal to $R_{cr}$ for an interaction involving a ground-state absorption by an acceptor ion, energy transfer can occur. Interactions involving excited-state acceptor ions can usually be neglected because at pump powers of a few Watts, the average separation between these excited-state ions is usually much larger than $R_{cr}$.

It is for these reasons that the cross-relaxation pathways illustrated in Figure 1 for a singly doped Tm$^{3+}$ system are the only ones that are significant. Both $C_1$ and $C_2$ involve interactions between sensitizer ions excited by the pump and acceptor ions in the ground state. However, there will be no energy transfer or radiation if multi-phonon relaxation is too rapid, which is the case in many crystals that have relatively high lattice phonon energies.

**Low phonon energy crystals**

Reducing the multi-phonon relaxation rates in crystalline hosts is accomplished by incorporating heavier halides, such as chlorine or bromine, which has the effect of reducing the maximum phonon energies in the crystal. As a result, electronic states with small energy gaps, corresponding to mid-IR transitions, that are normally quenched in conventional oxide and fluoride hosts will radiate in a chloride host. This effect has been used to build mid-IR rare earth-based solid-state lasers. For example, Pr$^{3+}$:LaCl$_3$ lasers have produced 5.2-μm [11] and 7.2-μm [12] emission. The LaCl$_3$ host is extremely hygroscopic and offers poor mechanical stability. However, lead salts offer better mechanical stability and moisture resistance and, when created with chlorine or bromine as the halide, also have low phonon energies. For example, a room temperature 4.6-μm erbium laser using KPb$_2$Cl$_5$ as the crystalline host and no environmental precautions to limit exposure to moisture has been demonstrated [13]. The KPb$_2$Cl$_5$ host has also been used to demonstrate a Dy$^{3+}$ 2.43-μm laser [14-16]. The success of infrared lasers using KPb$_2$Cl$_5$ as a host material has motivated further spectroscopic studies of Er$^{3+}$:KPb$_2$Cl$_5$ [17,18] in addition to other rare earth ions such as Pr$^{3+}$ [19,20] and Nd$^{3+}$ [21-24]. Activation of mid-infrared transitions of rare earth ions by reducing the phonon energies has been pushed further using KPb$_2$Br$_5$ as a host crystal [25,26]. This material has even lower phonon energies than KPb$_2$Cl$_5$ because of the substitution of Cl with the heavier Br.

**Crystal growth**

Crystals with heavy halides such as chlorine have low melting points. For LaCl$_3$, the melting point is 858°C; for KPb$_2$Cl$_5$, the melting point is 434°C; and for YCl$_3$, a host crystal used in a study of cross-relaxation of singly doped thulium crystals, the melting point is 721°C. The low melting point of all these crystals allows them to be grown in fused silica ampoules in a furnace constructed of fused silica with nickel-chromium resistance wire for heating. A self-seeded vertical Bridgman can be used to grow chloride crystals from melts of anhydrous-powered starting materials under a low-pressure (approximately 100 Torr) Cl$_2$ atmosphere, which is necessary to prevent the chloride compounds from disassociating. Methods for producing crystalline KPb$_2$Cl$_5$ and a documentation of its basic properties were reported in 1995 by Nitsch et al. [27]. Interest in incorporating rare earth ions into KPb$_2$Cl$_5$ has led to further refinements of material preparation and crystal growth techniques [28-31].

Data discussed in this paper are from rare earth ions doped into two different low phonon energy crystalline hosts. YCl$_3$ was chosen as a host to study Tm$^{3+}$ cross-relaxation because TmCl$_3$ and YCl$_3$ share the same monoclinic crystal structure. As a result, Tm$^{3+}$ ions incorporate at any concentration and occupy a single, highly symmetric site, which enables long excited-state lifetimes and a Stark structure that is partially resolvable even at room temperature. The KPb$_2$Cl$_5$ host was chosen to study singly doped crystals with Tm$^{3+}$ or Pr$^{3+}$ and a co-doped crystal with Tm$^{3+}$ and Pr$^{3+}$ because the crystal is stable under normal atmospheric conditions. In contrast, YCl$_3$ crystals will dissolve in a matter of minutes when exposed to normal atmospheric humidity. For the crystal growth of KPb$_2$Cl$_5$, stoichiometric mixtures were prepared from anhydrous, high-purity KCl and PbCl$_2$ powders mixed with small amounts of anhydrous, high-purity powders of PrCl$_3$ or TmCl$_3$, or both. To grow YCl$_3$, anhydrous, high-purity powdered YCl$_3$ and TmCl$_3$ were mixed. In all cases, the powdered mixtures were melted and allowed to sit molten under approximately 100 Torr of Cl$_2$ for several hours to reduce oxide impurities. The melt, contained in a 10-mm inner diameter fused silica ampoule with a tapered tip, was cooled over a period of 5 days while remaining under the Cl$_2$ atmosphere. The finished samples were polycrystalline with large grains and were un-oriented.
Spectroscopy

Unpolarized fluorescence spectra between 1,600 and 5,500 nm were collected with a 0.20-m monochrometer. Fluorescence was induced with laser diodes gated to produce 50-ms pulses. The diode pump powers were between 0.25 and 2.0 W. A pulse repetition rate of 10 Hz was used to synchronize a lock-in amplifier that received its input from a photo-detector mounted at the exit slits of the monochrometer. Spectra were collected using three passes - one for the 1,100- to 1,700-nm range, one for the 1,550- to 3,000-nm range, and one for the 3,000- to 5,500-nm range. An InGaAs photo-detector was used for the 1,100- to 1,700-nm range. For the other two spectral ranges that covered 1,550 to 5,500 nm, a liquid nitrogen-cooled InSb was used for photo-detection. For the 3,000- to 5,500-nm range, a long pass filter that blocked wavelengths less than 2,500 nm was in place to eliminate the short wavelength features from appearing in higher order. Also, for spectral acquisition at wavelengths greater than 2,500 nm, the monochrometer was purged with dry nitrogen gas in order to reduce a strong absorption feature at 4,300 nm resulting from atmospheric CO₂. Emission was measured with the Tm³⁺:YCl₃ remaining sealed in the fused silica ampoules to prevent degradation from exposure to atmospheric moisture. Fused silica is transparent for the range of emission wavelengths studied. For Tm³⁺:KPb₂Cl₅, no environmental precautions were used. In each case, the wavelength dependence of the complete light collection and detection system was calibrated using a blackbody source. Spectra were corrected using the system response function obtained from the blackbody calibration. During fluorescent decay measurements, the monochrometer acted as a filter to isolate emission at wavelengths associated with specific energy levels.

Results and discussion

Spectroscopy of singly doped Tm³⁺ crystals

Figure 2 shows a fluorescence spectrum at 300 K between 1,100 and 2,000 nm of Tm³⁺:KPb₂Cl₅ that results from pumping with an 805-nm laser diode [32]. The spectrum has three features that are typical of Tm³⁺ spectra in low phonon energy hosts. The emissions originate from the first three excited states and arise from the following transitions: ³H₄ → ³F₄ (centred at 1,450 nm), ³H₅ → ³H₆ (centred at 1,250 nm), and ³F₃ → ³H₆ (centred at 1,850 nm). The emissions at 1,450 and 1,250 nm are a characteristic of Tm³⁺ in a low phonon energy host crystal. For Tm³⁺ in YAG or YLF, these emissions are quenched by multi-phonon relaxation, and the only IR emission observed is the broadband centred at 1,850 nm arising from the ³F₄ level.

For the same 1,100- to 2,000-nm spectral range, a fluorescence spectrum from Tm³⁺:YCl₃ at 300 K arising from pumping with a 0.35-W, 811-nm laser diode is shown in Figure 3 [33]. Because YCl₃ is also a low phonon energy host, the same three spectral features appear. In YCl₃, the Tm³⁺ ions are at sites with higher symmetry than in KPb₂Cl₅. As a result, more of the Stark multiplet structure is resolvable in the emission lines in Figure 3 than in Figure 2. Also shown in Figure 3 is an overlap of a fluorescence spectrum at 400 K from the same crystal under the same pump conditions. As the temperature is increased, there is a small reduction in emission at 1,850 nm from the ³F₄ level, but a doubling in 1,250-nm emission from the ³H₅ level. The increase in emission from the ³H₅ as the temperature is raised is an interesting and counterintuitive result. The effect of a temperature increase on this emission is illustrated more graphically in Figure 4 [33]. It shows the normalized fluorescence intensity at three specific wavelengths as a function of temperature between 300 and 500 K. The wavelengths chosen reflect the populations of the first three excited states for Tm³⁺. The data show that the population of ³H₅ state increases relative to the other states as the temperature rises.
Cross-relaxation in singly doped Tm$^{3+}$ crystals

The anomalous behaviour of the 1,200-nm fluorescence from the $^3H_5$ state can be explained as arising from phonon-assisted cross-relaxation [34]. The processes illustrated in Figure 1 labelled C1 and C2 are both non-resonant and require phonon assistance to complete. C1 is the process already known in Tm$^{3+}$-doped YAG and YLF that involves an interaction between a $^3H_4$ ion activated by the pump and a $^3H_6$ ion in the ground state to produce two $^3F_4$ ions. The C1 process results in an excess of energy that must be converted to phonons. The energy gap for this cross-relaxation is 698 cm$^{-1}$, which corresponds to the creation of about three phonons in the YCl$_3$ lattice that has a maximum phonon energy of 260 cm$^{-1}$.

For the process C2 that feeds both the $^3F_4$ and $^3H_5$ levels, the energy gap is a deficit of $-641$ cm$^{-1}$. This process must absorb three phonons from the lattice to complete. However, phonon absorption processes have much stronger temperature dependence than phonon-emitting processes. At low temperatures, any relaxation process that emits phonons, such as cross-relaxation or multi-phonon relaxation, can proceed through spontaneous emission. At high temperatures, stimulated emission will occur as phonon occupation increases, which increases the relaxation rate. Therefore, the temperature dependence of the rate for a phonon emission process $W_e$ is given by

$$W_e(T) = W_e(T = 0)(1 + n)^{N_e},$$

where $N_e$ is the number of phonons ($\Delta E/\hbar \omega$) emitted to fill the energy gap $\Delta E$ that have energy $\hbar \omega$ and $n$ is the phonon occupation number [35].
However, phonon absorption processes must have occupied phonon states in order to proceed. The temperature dependence of the rate $W_a$ for a phonon absorption process is given by

$$W_a(T) = W_a(T=0)(n)^N_a,$$

(5)

where $N_a$ is the number of phonons absorbed. The temperature dependencies of Equations 4 and 5 arise because the phonon occupation number $n$ follows a Bose-Einstein distribution given by

$$n = (\exp[\hbar \omega / kT] - 1)^{-1},$$

(6)

where $\hbar \omega$ is the maximum phonon energy (260 cm$^{-1}$ for YCl$_3$) [36]. Therefore, the maximum phonon energy is the most important parameter in controlling the temperature and energy gap dependence of all phonon-assisted relaxation processes, including cross-relaxation and multi-phonon relaxation.

Excited state populations and lifetimes for Tm$^{3+}$, which ensue after pumping the $^3\text{H}_4$ state at 800 nm, depend on the competition between spontaneous emissions of radiation, cross-relaxation, multi-phonon relaxation, and up-conversion. At temperatures greater than 500 K, multi-phonon relaxation is the dominant process, which results in quenching of the fluorescence from all levels. At room temperature, near 300 K, multi-phonon relaxation is reduced and cross-relaxation can proceed. However, at 300 K, the occupation of phonon states is still substantial, which allows the endothermic process $C_2$ to compete with the exothermic process $C_1$.

A macroscopic model of the populations of the four lowest levels of Tm$^{3+}$ was constructed using coupled time-dependent rate equations [33]. Rate constants for spontaneous emission, cross-relaxation, and up-conversion were determined by fitting the model to fluorescence lifetime data at 300 K, a temperature at which multphonon relaxation can be neglected. Rate constants for multi-phonon relaxation were determined by fitting the model to lifetime data above 400 K, temperatures at which multi-phonon relaxation is significant [33].

After the determination of the rate constants, the relative populations of the three lower levels of Tm$^{3+}$ as a function of temperature were determined using the rate model and the temperature dependencies of Equations 4 and 5 for cross-relaxation processes $C_1$ and $C_2$, respectively. Figure 5 shows an overlay of the temperature-dependent rate modelling with the temperature-dependent intensity data from Figure 4 [33]. The model predicts the observed increase in emission from the $^3\text{H}_4$ level as the temperature is raised. The model shows that the branching ratio for the $^3\text{H}_4$ to $^3\text{H}_5$ transition is less than 1%, and as a result, the population of the $^3\text{H}_5$ arises almost entirely from the $C_2$ cross-relaxation process [33]. Between 300 and 400 K the model also predicts the observation that the emission from the $^3\text{F}_4$ and $^3\text{H}_4$ levels is unchanged as the temperature rises because multi-phonon relaxation has not increased to a level that it competes with radiation and cross-relaxation.

This result is significant because it implies that the process $C_2$ converts lattice phonons into 1,200-nm radiation, which is a cooling effect. In contrast to previous demonstrations of solid-state optical cooling from anti-Stokes emission [37-43], cooling from cross-relaxation will not lose efficiency at low temperatures because the $-641\text{ cm}^{-1}$ energy gap for the process is temperature independent. At low-temperatures, cooling from anti-Stokes emission loses efficiency because of thermal depopulation of the upper Stark levels.

Also of interest for Tm$^{3+}$:YCl$_3$ is that additional study of the concentration dependence of the cross-relaxation rates determined that the critical radius $R_c$ at room temperature for the energy transfer is about 15 Å. That distance is comparable to $R_c$ for Tm$^{3+}$ cross-relaxation.
in conventional oxide and fluoride hosts [7,8]. This implies that the endothermic cross-relaxation process $C_2$ is enabled by the reduction in multi-phonon quenching and not because interaction rates between neighbouring $\text{Tm}^{3+}$ ions are changed significantly by a chloride host. These spectroscopic results suggest that a heat generation study should be conducted for the near-IR-pumped $\text{Tm}^{3+}$ in a low phonon energy host.

**Energy transfer in $\text{Tm}^{3+}$-$\text{Pr}^{3+}$ co-doped crystals**

In addition to its own IR-emitting properties, the $\text{Tm}^{3+}$ ion has been used to sensitize other rare earth ions for diode pumping. Most notable is the $\text{Ho}^{3+}$ ion, which has a useful IR laser transition at 2.1 $\mu$m from its first excited state to its ground state but lacks a level that absorbs at 800 nm. Energy transfer from $\text{Tm}^{3+}$ to $\text{Ho}^{3+}$ has been used to create diode-pumped 2.1-$\mu$m lasers using YLF [7] and YAG [8] host crystals. $\text{Tm}^{3+}$ sensitization has also been used in low phonon energy crystals. For example, $\text{Nd}^{3+}$ has absorption near 800 nm. However, for co-doped $\text{Tm}^{3+}$-$\text{Nd}^{3+}$:KPb$_2$Cl$_5$, the presence of the $\text{Tm}^{3+}$ is known to increase the absorption of the pump and enhance the IR emission from the $\text{Nd}^{3+}$ ions [44]. An additional example is co-doped $\text{Tm}^{3+}$-$\text{Pr}^{3+}$:CsCdBr$_3$, in which pumping the $3\text{H}_4$ level of $\text{Tm}^{3+}$ results in energy transfer and up-conversion to emitting states in the visible [45].

Energy transfer from the $3\text{H}_4$ state of $\text{Tm}^{3+}$ to the IR-emitting states of $\text{Pr}^{3+}$ in a low phonon energy host crystal is also an interesting phenomenon. Like $\text{Ho}^{3+}$, the $\text{Pr}^{3+}$ ion also lacks absorption at 800 nm. However, transitions out of the first three excited states of $\text{Tm}^{3+}$ that populate through cross-relaxation are resonant with absorption transitions out of the $\text{Pr}^{3+}$ ground state to excited states of $\text{Pr}^{3+}$ that radiate in the mid-IR. Figure 6 compares the lower energy levels of $\text{Tm}^{3+}$ to the lower levels of $\text{Pr}^{3+}$ and illustrates three possible pathways for resonant energy transfer that involve excited-state $\text{Tm}^{3+}$-sensitizing ions interacting with ground-state $\text{Pr}^{3+}$ acceptor ions.

In contrast to $\text{Pr}^{3+}$:YAG or $\text{Pr}^{3+}$:YLF, $\text{Pr}^{3+}$ ions in a chloride host crystal will radiate at mid-IR wavelengths because the lower energy levels are no longer quenched by multi-phonon relaxation. This effect was exploited to make 5.2- and 7.2-$\mu$m lasers using $\text{Pr}^{3+}$:LCl$_3$ [11,12]. For $\text{Pr}^{3+}$ doped into KPb$_2$Cl$_5$, the lower energy levels will also radiate in the mid-IR. The mid-IR fluorescence can be observed in singly doped $\text{Pr}^{3+}$:KPb$_2$Cl$_5$ under this pump, the room temperature fluorescence that results from 1,600 to 2,800 nm is shown in Figure 7 and from 3,000 to 5,500 nm is shown in Figure 8 [32]. Each feature in the spectra is labelled with the associated $\text{Pr}^{3+}$ energy level transition.

The $\text{Tm}^{3+}$ sensitization of $\text{Pr}^{3+}$:KPb$_2$Cl$_5$ allows for more convenient 800-nm diode pumping. For a co-doped $\text{Tm}^{3+}$-$\text{Pr}^{3+}$:KPb$_2$Cl$_5$ crystal using a 1.5-W, 805-nm laser diode as a pump source, the same broadband mid-IR emission between 4,000 and 5,500 nm from the $\text{Pr}^{3+}$ ions is observed. The room temperature fluorescence that results from 805-nm pumping of the co-doped crystal overlapped with the fluorescence that results from the 1,483-nm pumping of the same co-doped crystal from 1,600 to 2,800 nm is shown in Figure 9 and from 3,000 to 5,500 nm is shown in Figure 10 [32]. The overlap of the two spectra in Figure 10 show that the broadband emission between 4,000 and 5,500 nm from the $\text{Pr}^{3+}$ ions is essentially the same for the two crystals, even

---

**Figure 6 Energy transfer processes for co-doped $\text{Tm}^{3+}$-$\text{Pr}^{3+}$:KPb$_2$Cl$_5$.** The first three excited states of $\text{Tm}^{3+}$-sensitizing ions are all resonant with ground-state transitions of $\text{Pr}^{3+}$ acceptor ions.
though the pump sources are at different wavelengths. As expected, Figure 9 shows less emission at 2,400 nm out of the $^3F_4$ state of Pr$^{3+}$ in the co-doped crystal compared to 1,483-nm pumping of the singly doped crystal, because the $^3F_4$ state of Pr$^{3+}$ is no longer being pumped directly.

Figure 11 shows lifetime data for the 1,450-nm emission from the $^3H_4$ level of Tm$^{3+}$ resulting from 805-nm pumping for the singly doped and co-doped samples [32]. Comparison of the 1,450-nm emission from Tm$^{3+}$:Kpb$_2$Cl$_5$ to 1,450-nm emission from Tm$^{3+}$:Pr$^{3+}$:Kpb$_2$Cl$_5$ shows the rapid quenching of emission from the Tm$^{3+}$ because of energy transfer to the Pr$^{3+}$. Analyses of the Tm$^{3+}$ lifetime data for the co-doped crystal show that the energy transfer processes from the Tm$^{3+}$ sensitizers to the Pr$^{3+}$ acceptors have high quantum efficiencies.

Figure 7 Fluorescence from 1,600 to 2,800 nm resulting from 1,483-nm pumping of Pr$^{3+}$:Kpb$_2$Cl$_5$. The sample has a Pr$^{3+}$ concentration of $1.5 \times 10^{20}$ ions/cm$^3$.

Figure 8 Fluorescence from 3,000 to 5,500 nm resulting from 1,483-nm pumping of Pr$^{3+}$:Kpb$_2$Cl$_5$. The sample has a Pr$^{3+}$ concentration of $1.5 \times 10^{20}$ ions/cm$^3$. 
For the energy transfer process labelled $T_1$ in Figure 6, the quantum efficiency $\eta_1$ is estimated at 94%, and for the process labelled $T_2$ in Figure 6, the estimated quantum efficiency $\eta_2$ is 83% [32]. The process labelled $T_3$ can be neglected because the $^3H_5$ level of Tm$^{3+}$ never obtains significant population. Further analysis of the decay transients provides estimates of 11 and 12 Å, respectively, for the critical radii of energy transfer from the $^3H_4$ and $^3F_4$ states of Tm$^{3+}$. The critical radii for this co-doped system are comparable to the critical radii of electric dipole-dipole interactions involving rare earth ions in other host crystals, such as the cross relaxation of Tm$^{3+}$ in YCl$_3$ discussed in the earlier part of this paper.

Figure 9 Fluorescence from 1,600 to 2,800 nm from Tm$^{3+}$-Pr$^{3+}$:K Pb$_2$Cl$_5$. Fluorescence resulting from 1,483-nm pumping of Tm$^{3+}$-Pr$^{3+}$:K Pb$_2$Cl$_5$ compared to fluorescence resulting from 805-nm pumping. The sample has a Pr$^{3+}$ concentration of $1.5 \times 10^{20}$ ions/cm$^3$ and a Tm$^{3+}$ concentration of $3.0 \times 10^{20}$ ions/cm$^3$.

Figure 10 Fluorescence from 3,000 to 5,500 nm from Tm$^{3+}$-Pr$^{3+}$:K Pb$_2$Cl$_5$. Fluorescence resulting from 1,483-nm pumping of Tm$^{3+}$-Pr$^{3+}$:K Pb$_2$Cl$_5$ compared to fluorescence resulting from 805-nm pumping. The sample has a Pr$^{3+}$ concentration of $1.5 \times 10^{20}$ ions/cm$^3$ and a Tm$^{3+}$ concentration of $3.0 \times 10^{20}$ ions/cm$^3$. 
The usefulness of this system is that it functions as an optically pumped mid-IR phosphor that converts light from 805-nm diodes to broadband mid-IR from 4 to 5.5 μm. The 805-nm diode sources are low in cost compared to 1.5- or 2-μm sources that would activate the Pr3+ mid-IR emission directly. This material could be used as a low-cost method for detecting gases with absorptions in the 4- to 5.5-μm range. For example, the small dip at 4,300 nm in the fluorescence spectra shown in Figure 10 is an artefact of a less than perfect purge of the optical path with dry nitrogen. Without the purge, the 4,300-nm fluorescence emitted by the diode-pumped crystal is completely absorbed by atmospheric CO2. In effect, the experimental setup functioned as a very sensitive atmospheric CO2 detector.

Conclusions

This paper discussed two applications of Tm3+ sensitization of rare earth-doped low phonon energy host crystals, in which the resulting reduction in multi-phonon relaxation rates enables useful energy transfer processes to occur that are quenched in conventional oxide and fluoride crystals. One application is the enabling of an endothermic cross-relaxation process for Tm3+ that converts lattice phonons to infrared emission near 1,200 nm. The existence of this process suggests that endothermic phonon-assisted energy transfer could be a fundamentally new way of achieving optical cooling in a solid. The other application is a novel optically pumped mid-IR phosphor that converts 805-nm light from readily available low-cost diodes into broadband emission from 4 to 5.5 μm. The phosphor is efficient, low-cost, and scalable.

Application of theories for electric dipole-dipole sensitizer-acceptor interactions shows that the critical radii for energy transfer processes between rare earth ions do not change significantly between various host crystals. The novel energy transfer processes observed in low phonon energy host crystals occur because the multi-phonon relaxation rates for the levels involved are reduced and no longer compete with the radiative and non-radiative energy transfer rates. In imagining new kinds of applications for low phonon energy crystals, circumstances in which the multi-phonon relaxation rates can be reduced to much less than the known rates for electric dipole interactions should be investigated.

Competing interests

The authors declare that they have no competing interests.

Authors’ contributions

JG drafted the manuscript, prepared the samples, and participated in acquiring, analyzing and interpreting the data and in conceiving and designing these experiments. SRB participated in acquiring, analyzing, and interpreting the data, in conceiving and designing these experiments, and in revising the manuscript. Both authors read and approved the final manuscript.

Acknowledgements

Work at Loyola University Maryland was supported by the National Science Foundation Division of Electrical and Communication Systems under grants ECS-9970055 and ECS-0245455. The Office of Naval Research supported this work at the Naval Research Laboratory.

Author details

1Department of Physics, Loyola University Maryland, 4901 N. Charles Street, Baltimore, MD 21210, USA. 2Optical Sciences Division, Naval Research Laboratory, 4555 Overlook Avenue SW, Washington, DC 20375, USA.
