Oxidation-Responsive, Tunable Growth Factor Delivery from Polyelectrolyte-Coated Implants

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Polyelectrolyte multilayer (PEM) coatings, constructed on the surfaces of tissue engineering scaffolds using layer-by-layer assembly (LbL), promote sustained release of therapeutic molecules and have enabled regeneration of large-scale, pre-clinical bone defects. However, these systems primarily rely on non-specific hydrolysis of PEM components to foster drug release, and their pre-determined drug delivery schedules potentially limit future translation into innately heterogeneous patient populations. To trigger therapeutic delivery directly in response to local environmental stimuli, an LbL-compatible polycation solely degraded by cell-generated reactive oxygen species (ROS) was synthesized. These thioketal-based polymers were selectively cleaved by physiologic doses of ROS, stably incorporated into PEM films alongside growth factors, and facilitated tunable release of therapeutic bone morphogenetic protein-2 (BMP-2) upon oxidation. These coatings' sensitivity to oxidation was also dependent on the polyanions used in film construction, providing a simple method for enhancing ROS-mediated protein delivery in vitro. Correspondingly, when implanted in critically-sized rat calvarial defects, the most sensitive ROS-responsive coatings generated a 50% increase in bone regeneration compared with less sensitive formulations and demonstrated a nearly threefold extension in BMP-2 delivery half-life over conventional hydrolytically-sensitive coatings. These combined results highlight the potential of environmentally-responsive PEM coatings as tunable drug delivery systems for regenerative medicine.

1. Introduction

Over half a million bone grafting procedures are performed annually in the United States,[1] and despite the resultant need for bone-substituting materials, regenerating large-scale orthopedic defects in the clinic remains challenging. The current gold-standard treatment involves harvesting autologous bone and surgically integrating the extracted tissue into the defect site; however, this approach is constrained by both donor-site morbidity and the limited amount of harvestable donor bone available in a given patient.[2] Accordingly, considerable research effort has been expended to develop generalizable, synthetic bone grafts comprising biodegradable scaffolds that locally deliver regenerative therapeutics. Natural bone healing relies on cell signaling pathways that are primarily directed by secreted growth factors, including the highly osteoinductive bone morphogenetic protein-2 (BMP-2).[3] BMP-2 has been identified as the most potent member of the BMP family due to its ability to promote cellular proliferation of progenitor cells and spur stem cell differentiation into bone-forming osteoblasts.[4] Moreover, this therapeutic protein has been heavily investigated in drug delivery applications,[5] and was introduced in 2002 in a United States Food and Drug Administration (FDA)-approved clinical formulation (INFUSE Bone Graft from Medtronic[6]) that releases BMP-2 from an absorbable collagen sponge for certain spinal, dental, and maxillofacial indications.

However, many of these biomaterial-based BMP-2 delivery systems discharge the majority of their drug payload in a short-duration bolus over the first several minutes to hours,[7] thereby necessitating that these materials contain high protein doses to compensate for the poorly-controlled release kinetics and rapid clearance.[8] The physiologically-inappropriate BMP-2 dosing facilitated by these carriers not only causes damaging side-effects,[9] but also prematurely depletes the encapsulated growth factor reservoir and can lead to incomplete bone regeneration.[10] Motivated by these persistent shortcomings, many groups have explored the impact of BMP-2 dosing[10] and release kinetics[11] from biomaterial implants alongside methodologies for tuning these parameters. In particular, polyelectrolyte multilayer (PEM)
thin film deposition represents a promising technology for facilitating localized, highly tunable therapeutic delivery from implanted biomaterials.\textsuperscript{[12]}

Previous work has demonstrated that PEM assemblies can form nanolayered surface coatings on a variety of scaffold materials and are easily fabricated under mild aqueous conditions from a layer-by-layer (LbL) iterative adsorption\textsuperscript{[13]} of charged species. Moreover, the incorporation of therapeutic compounds, including nucleic acids\textsuperscript{[14]} small molecules,\textsuperscript{[15]} or proteins\textsuperscript{[16]} into the PEM architecture alongside a biodegradable polymer allows for the pre-programmed release of drug cargo from the film coating\textsuperscript{[17]}. These films have primarily utilized hydrolytically-degradable, cationic poly(β-amino ester) (PBAE) polymers that break down over time in the presence of water via ester hydrolysis\textsuperscript{[18]} to discharge the film-encapsulated payloads. Modulation strategies for tuning PEM drug release kinetics include modifying the hydrophobicity of the PBAE,\textsuperscript{[19]} creating covalent cross-links between layers,\textsuperscript{[20]} incorporating barrier layers in the film architecture,\textsuperscript{[21]} or staggering depositions of different therapeutics to foster sequential release of multiple payloads.\textsuperscript{[22]} In our previous orthopedic work, we have demonstrated extended BMP-2 release from LbL thin film coatings to generate bone both around a non-degradable tibial implant\textsuperscript{[23]} and within a critically-sized cranial defect.\textsuperscript{[24]}

However, these hydrolytically degradable PEM systems are minimally-responsive to the local tissue environment and are primarily dependent on previously determined drug delivery kinetics for achieving tissue regeneration. Therefore, this work sought to develop a PEM coating that delivers therapeutics directly and selectively in response to local tissue repair signaling and healing-specific environmental stimuli. Many recent drug delivery technologies have employed materials that respond to external triggers, though most promising are “smart” systems\textsuperscript{[25]} that enable on-demand drug release in response to tissue-produced signals, including pH levels,\textsuperscript{[26]} enzymatic activity, or the presence of reactive oxygen species (ROS). ROS are particularly attractive for facilitating material responsiveness since these cell-specific stimuli are highly associated with bone growth and remodeling. ROS are important mediators in many biological processes and the body’s immune response,\textsuperscript{[27]} though ROS levels are also elevated across a number of pathologies\textsuperscript{[28]} and at sites of inflammation and tissue injury.\textsuperscript{[29]} In orthopedics specifically, mature osteocytes actively remodeling bone tissue have been shown to generate increased ROS as part of the natural healing cycle,\textsuperscript{[30]} further motivating the targeting of ROS as a selective mediator of therapeutic release from PEM-coated implants.

Several polymer-based technologies that utilize cellular oxidation as a selective trigger for biomaterial responsiveness have recently emerged,\textsuperscript{[31]} including ROS-degradable tissue engineering constructs\textsuperscript{[32]} and oxidation-sensitive drug delivery vehicles. These triggerable therapeutic carriers include nanoparticles,\textsuperscript{[33]} microparticles,\textsuperscript{[34]} and hydrogels\textsuperscript{[35-37]} that specifically deliver payloads to intracellular or extracellular targets featuring high levels of oxidative stress. Though some groups have employed positively charged, oxidation-sensitive polymers for nanoparticle fabrication,\textsuperscript{[34]} to our knowledge ROS-degradable polymers have never been employed in the construction of LbL coatings. To confer ROS-responsiveness to different biomaterials, many groups have employed thioketal-based polymers due to their ease of synthesis,\textsuperscript{[35]} sensitivity to many different ROS,\textsuperscript{[36]} and oxidation-specific degradation.\textsuperscript{[31b]} Herein, we report the development of a thioketal-based polycation that forms stable PEM coatings with growth factor proteins upon LbL assembly, selectively releases these therapeutics upon oxidation, and features tunable sensitivity to ROS-mediated protein delivery. Most importantly, the oxidation-responsive drug release kinetics from PEM-coated implants directly correlate with bone growth outcomes in critically-sized calvarial defects. These findings highlight the utility of environmentally-responsive therapeutic delivery in orthopedic regeneration, and the potential to incorporate this principle for LbL-generated delivery systems.

2. Results and Discussion

2.1. Synthesis and Characterization of an ROS-Degradable, LbL-Compatible Polycation

To synthesize the LbL-compatible poly(thioketal β-amino amide) (PTK-BAA) polycation, the terminal amine group in a cysteamine monomer was successfully protected with ethyl trifluoroacetate before further reacting with 2,2-dimethoxypropane to form a protected thioketal diamine monomer as demonstrated in Figure 1A. Following deprotection, the thioketal diamine was further functionalized with reactive acrylamide groups and then polymerized via Michael addition condensation with 4,4′-trimethylene dipiperidine to generate the PTK-BAA polycation (Figure 1A). A conventional PBAE polymer was similarly synthesized as shown in Figure 1B. These Michael addition polymerizations leave both polycations with ionizable tertiary amines in the polymer backbone, though the PTK-BAA replaces the PBAE’s hydrolytically-cleaved esters with hydrolytically-inert amides and introduces ROS-sensitive thioketal units in the main polymer sequence to channel chain scission exclusively through oxidative degradation.

The PTK-BAA structure was confirmed with \textsuperscript{1}H NMR (Figure 1C), while MALDI-TOF analysis also confirmed the presence of predicted polymerization products and established the bulk polymer’s number average molecular weight (\(M_n\)) as 2267 g mol\(^{-1}\) with a polydispersity index (PDI) of 1.36 (Figure 1D). \textsuperscript{1}H NMR also indicated the successful formation of the PBAE polycation, while GPC analysis showed a polymer \(M_n\) of 6621 g mol\(^{-1}\) with a PDI of 1.48. As indicated in previous reports,\textsuperscript{[34]} the PTK-BAA Michael addition condensation of diacrylamide monomers yielded relatively low molecular weight polymers compared with PBAEs polymerized with diacrylate precursors (\(\approx 2.3\) kDa vs \(\approx 6.6\) kDa). The diacrylamide polymerization, carried out for 24 h at room temperature, was also relatively insensitive to longer reaction times or higher temperatures while converting to the highest molecular weight when synthesized in a solvent mix of ethanol and water,\textsuperscript{[36]} instead of conventionally used tetrahydrofuran (THF) or dimethylformamide (DMF). To serve as a direct non-degradable control for the PTK-BAA, the thioketal-less polycation hexane-poly(β-amino amide) (hexane-PBAA) was also successfully synthesized (Figure S1, Supporting Information).

To demonstrate the selective ROS-sensitivity of the thioketal-containing PTK-BAA polycations, polymer samples were incubated in escalating doses of hydrogen peroxide (H\(_2\)O\(_2\)) and evaluated for thioketal degradation by \textsuperscript{1}H NMR. As shown...
Figure 1. Synthetic scheme for LbL-compatible polycations, A) the oxidation-sensitive poly(thioketal β-aminoamide) (PTK-BAA) and B) the hydrolytically-degradable poly(β-aminoester) (PBAE). Both polycations are polymerized using Michael addition condensation and contain ionizable tertiary amines in the polymer backbone. However, the PTK-BAA features ROS-sensitive thioketal units and minimally-degradable amide linkers while the PBAE is degraded through the hydrolytically-labile ester bonds. Successful PTK-BAA polymer synthesis was confirmed by C) 1H NMR spectroscopy and by D) MALDI-TOF mass spectrometry. The presence of predicted polymerization products in the polymer bulk further validates the successful step-growth synthesis.

Figure 2. In vitro PTK-BAA degradation. Polymer samples were incubated for 72 h in escalating doses of H₂O₂ and evaluated by A) NMR, indicating that the characteristic thioketal peak underwent dose-dependent degradation. B) Polymers samples incubated in 10 mm H₂O₂ and evaluated by MALDI-TOF also demonstrated the complete disappearance of higher molecular weight signal and generated species corresponding with predicted polymer degradation products. Selective thioketal sensitivity to oxidation is well documented, though encouragingly PTK polymer degradation could be further amplified in vivo by more highly reactive compounds (particularly hydroxyl radicals) that are commonly produced in regenerating tissues. Cytotoxicity of the PTK-BAA polymer and its degradation products with MC3T3-E1 pre-osteoblast cells was also determined in comparison to no treatment, treatment with a cytotoxic control 25kDa LPEI polymer, and treatment with the conventional PBAE polycation used in previous PEM film formulations (Figure S2, Supporting Information). As measured by the Cell Titer Glo assay of relative viable cell number, all treatments exhibited significantly higher viability than LPEI treatment as expected while cells incubated with the parent and degraded PTK-BAA polymers were not statistically different from no treatment. Though these results only featured cells treated with film constituents rather than intact films, previous work conclusively demonstrated robust cell attachment to
Figure 3. Characterization of PTK-BAA films fabricated with BMP-2 and 450 kDa PAA. A) Profilometry measurements of film thickness versus number of tetralayer depositions indicate controlled, linear film growth. Tunability of total BMP-2 loading in PTK-BAA films via number of tetralayer depositions is demonstrated by B) IVIS imaging of Cy5-labeled BMP-2, and C) ELISA measurements of total BMP-2 recovered from degraded films.

2.2. LbL Film Construction, Characterization, and In Vitro Drug Release

To first demonstrate the LbL-compatibility of PTK-BAA polycations, model silicon substrates were alternatively dipped in aqueous baths of polyelectrolytes at pH 5.0 to form repeating tetralayer units. The initial layer was the ROS-degradable PTK-BAA polycation, which was followed by the polyanion poly(acrylic acid) (PAA, 450 kDa), the cationic model protein lysozyme (fluorescently tagged for quantitation), and then finally PAA again; these tetralayer cycles were repeated 30 times to construct the complete LbL films on model silicon substrates as shown in Figure S3A, Supporting Information. PAA is minimally cytotoxic and effectively sequesters cationic growth factors into PEM films due to its high density of ionizable carboxylate groups,[22b] while the highly charged lysozyme (isoelectric point 11.35) has been previously employed as a low-cost model protein for encapsulation studies in LbL assembly.[16a,38] Film-coated silicon samples were incubated in phosphate buffered saline (PBS, pH 7.4) with escalating quantities of hydrogen peroxide and demonstrated significant dose-dependent protein release as shown in Figure S3B, Supporting Information. A minimal amount of lysozyme is released during incubation in pure PBS, likely initiated by weakened ionic bonds and film swelling during the transition from pH 5.0 to 7.4,[19] the substantial increase in protein liberation seen with lysozyme treatment strongly implicates oxidation as the chief driver of drug delivery from PTK-BAA films.

With ROS-triggered protein delivery from PEM constructs successfully validated, PTK-BAA film coatings were fabricated with the therapeutic growth factor protein BMP-2 and characterized. PEM films created on silicon with PTK-BAA, 450 kDa PAA, and BMP-2 displayed a linear increase in film thickness as a function of tetralayer depositions (Figure 3A), indicating robust LbL assembly. Next, films were similarly constructed with fluorescently tagged BMP-2 on 1 mm-thick, microporous poly(lactic-co-glycolic) acid (PLGA) scaffolds fabricated by solvent phase-separation.[22a] Importantly, the LbL dip-coating process formed PEM films on both the top and bottom PLGA surfaces to double the allowable drug loading per sample, and measured BMP-2 loadings (totaled from both sides of each coated scaffold, giving a cumulative surface area of \( \approx 1 \text{ cm}^2 \) for an 8.0 mm diameter sample) displayed a linear increase with increasing tetralayer additions (Figure 3B,C). This tunable therapeutic encapsulation is therefore easily controlled by varying the number of layer depositions to target a specific drug dosing per surface area of coated substrate.

In vitro BMP-2 release kinetics from the oxidation-sensitive films were next determined and compared against drug discharge from conventional hydrolytically-degradable PEM films fabricated with PBAE polycations. Film-coated PLGA scaffolds (30 tetralayers) were incubated in PBS or 1 mM H\(_2\)O\(_2\)-doped PBS and evaluated for BMP-2 release into the supernatant. As shown in Figure 4A, PBAE film coatings released the majority of their complexed growth factor over the first 24 h and demonstrated no difference in release kinetics between PBS or ROS-treated samples. However, PTK-BAA film coatings released relatively little BMP-2 in saline alone but expelled significantly more protein when treated with 1 mM H\(_2\)O\(_2\) (Figure 4A). Polyester-based materials have previously demonstrated minimal responsiveness to ROS-mediated degradation,[31b] and the exhibited initial bolus release of growth factor from PEM films is in line with prior reports.[22a]
To further assess the ROS-responsiveness of the PTK-BAA/BMP-2 coatings, drug release from similarly fabricated samples was evaluated with either a “constant” 1 mM H₂O₂ medium or with a “pulsed” medium that was switched back and forth between PBS and 1 mM H₂O₂ every 4 days as shown in Figure 4B. Following the initial 24 h burst release, samples incubated in the “constant” ROS media released significantly more BMP-2 than the “pulsed” samples incubated in PBS over the first 4-day cycle. However, upon switching the “pulsed” medium to 1 mM H₂O₂ over days 5–8, the release kinetics were statistically indifferent between the two sample sets (Figure 4B). The oscillating protein release behavior from “pulsed” samples was generally continued until day 20, though with some hysteresis in the responsiveness; the expected PBS “pulse” response during days 9–12 was delayed by 2 days, while the anticipated ROS media response during days 13–16 was also shortly postponed as shown in Figure 4B. Interestingly, at later time points the pulsed samples do not deliver extra BMP-2 despite possessing more of their initial drug reservoir, though this can potentially be explained by the pulsed samples simply not releasing their entire payload over 20 days due to the reduced oxidative stimulation. Additionally, PEM interlayer diffusion and film rearrangement that could occur in the non-oxidative cycles may have caused repartitioning of BMP-2 in the film such that less protein was readily exposed when the PTK-BAA component was degraded in ROS media. While this experiment confirmed triggerable BMP-2 delivery based on a global stimulus, these data also indicate this system’s potential for selective drug release to areas of cellular growth and tissue regeneration. Bone remodeling osteoclasts have been previously shown to localize at the periphery of bone injuries and produce high levels of ROS during the healing process. With PTK-BAA coated implants, osteogenic cellular activity could directly liberate BMP-2 at the regenerative defect margin, thereby limiting off-target effects, while preserving the implant’s undisturbed drug reservoir until sufficient new tissue growth reaches the defect’s central regions and releases more growth factor.

2.3. In Vitro Bioactivity of Film-Released BMP-2

The bioactivity of BMP-2 released from PTK-BAA films was evaluated by incubating MC3T3-E1 pre-osteoblasts with releaseate from PBS or 1 mM H₂O₂-treated film samples. BMP-2-containing releaseate from 30 tetralayer films constructed with PTK-BAA and 450 kDa PAA was first combined into discrete pools across the time course, and then a constant volume of releaseate from each pooled sample (to normalize the time-dependent dose of BMP-2) was lyophilized for 48 h to both concentrate the growth factor and remove residual cytotoxic peroxide as previously reported. The dried releaseate was then re-suspended into cell culture media and treated onto the pre-osteoblasts for 48 h before allowing further osteogenic cellular differentiation over an additional 4 days. Alkaline phosphatase production by cells was then quantified after normalization to total cellular protein levels, and as demonstrated in Figure 5, confirmed the bioactivity of film-released BMP-2. Moreover, cells treated with releaseate from ROS-treated films significantly increased their alkaline phosphatase levels compared with cells incubated in PBS releaseate, particularly at the later time points despite some variability potentially caused by osmotic stress on the cells from residual releaseate salts (Figure 5). These data further validate the connection between ROS-mediated drug release and regenerative cellular activity.

2.4. Effects of PEM Film Components on In Vitro BMP-2 Release

While the results discussed above establish the initial validation of oxidation-sensitive PEM coatings for responsive therapeutic delivery, interestingly the ROS-mediated discharge of lysozyme (Figure S3, Supporting Information) and BMP-2 (Figure 4A) were markedly different despite similar film formulations. It was hypothesized that a faster-releasing lysozyme’s lower molecular weight (14.3 vs 26 kDa for BMP-2) was creating less stable LbL assemblies that were more prone to disassociation.
Figure 5. In vitro cellular osteogenesis corresponds with oxidation-mediated BMP-2 release from PTK-BAA films. MC3T3-E1 pre-osteoblasts treated with releaseate from 1 mM H₂O₂-incubated films generated significantly more alkaline phosphatase than cells treated with releaseate from PBS-incubated films (n = 3 per treatment, *p < 0.05). 30 tetralayer films constructed with 450 kDa PAA.

Upon ROS-mediated PTK-BAA degradation. Previous work has demonstrated that lower molecular weight polyelectrolytes can more easily diffuse through a multilayer assembly; since the BMP-2 size is not easily alterable and the PTK-BAA already features a relatively low molecular weight (≤2.3 kDa), the impact of varyingly sized PAA polyanions in PEM films was explored to further adapt the LbL assembly stability. PTK-BAA/BMP-2 films were constructed with 450, 5, or 1.8 kDa PAA on PLGA scaffolds, and crucially did not display appreciable differences in total drug loading across formulations (Figure 6A). Additionally, all three formulations displayed limited protein release under non-oxidative conditions while delivering significantly more BMP-2 after treatment with 1 mM H₂O₂ as expected (Figure 6B). Further validating PTK-BAA films’ fundamental ROS-specific release mechanism, BMP-2 films constructed with hexane-PBAA (no ROS-degradable groups) or conventional PBAE polycations did not significantly increase their drug release in oxidative conditions (Figure S4, Supporting Information).

Although ROS-mediated BMP-2 release from 450 and 5 kDa PAA films was nearly indistinguishable in oxidative media, the 1.8 kDa PAA samples displayed a comparatively significant increase (greater than 40%) in drug release at both 1 and 0.1 mM H₂O₂ (Figure 6B). These BMP-2 liberation kinetics reasonably support the initial hypothesis that reducing PEM component molecular weight will decrease overall film stability and increase film disassociation (e.g., BMP-2 release) upon PTK-BAA degradation. However, these findings do not generate a strict molecular weight-dependent pattern given that films constructed with 5 kDa PAA behave more similarly to those with 450 kDa PAA than films containing 1.8 kDa PAA. Preliminary investigations also explored films made with 50 kDa PAA and found they did not substantially differ from those made with 450 or 5 kDa PAA (data not shown), further separating the performance of the 1.8 kDa formulation.

A possible explanation arises from the relationship between the PTK-BAA polycation and PAA polyanion; when the PAA’s molecular weight exceeds that of the degradable polymer, the highly-charged yet non-degradable PAA fosters more robust film stability. Conversely, if the PAA component’s molecular weight approaches or falls below that of the PTK-BAA (1.8 < 2.3 kDa), the degradable polycation chains more strongly impact the film architecture and confer additional sensitivity to ROS-mediated multilayer disassembly. Though LbL assemblies are considered effective ionic networks with significant amounts of interpenetration and entanglement, here it would appear that the network stability is highly dependent on the higher molecular weight component in the film. It should also be noted that while this work sought to improve sensitivity to a specific film-disrupting stimulus, many drug-releasing PEM assemblies are designed to maximize film stability to prolong delivery windows. However, for this environmentally-responsive system, these findings present an additional strategy for enhancing stimuli-triggered therapeutic delivery from PEM assemblies.

2.5. In Vivo Bone Regeneration and Drug Release Kinetics from PTK-BAA/BMP-2 Coated Implants

To explore the efficacy of ROS-responsive, PEM film-mediated growth factor delivery in vivo, PLGA scaffolds coated with PEM films comprising PTK-BAA, BMP-2, and the varyingly sized PAA polyanions (450, 5, or 1.8 kDa) were implanted in large-scale rat calvarial defects to evaluate in vivo bone regeneration. The surgically created 8.0 mm-diameter bone injuries are generally accepted as “critically-sized” in that the animal cannot completely heal over their natural lifetime and have been extensively used to evaluate regenerative biomaterial and bone tissue engineering strategies. As visualized in computed tomography (CT) micrographs in Figure 7A, new bone development (identified by its lower CT density compared with mature bone) is present at the defect edges for all three film formulations at week 4. This bone growth into the defect margins was quantified using an 8 mm-diameter region of interest (ROI) to selectively measure new bone as seen in Figure S5, Supporting Information. The 1.8 kDa PAA PEM coatings, featuring the highest sensitivity to ROS-mediated BMP-2 release (Figure 6B), promoted a greater than 50% increase in bone volume (Figure 7B) and significantly improved tissue mineralization (Figure 7C) within the defect margins when compared against the notably less responsive 450 and 5 kDa PAA film formulations. These significant results further supported the structure-function relationship between PEM components and BMP-2 release as explored in Figure 6. Moreover, histology of bone defects treated with the 1.8 kDa PAA films further demonstrated developing bone formation, though not fully mature bone coverage, in the calvarial defects at week 4 as shown in Figure S6, Supporting Information.

Importantly, the bone regeneration mediated by these oxidation-sensitive drug coatings indicated that tissue-generated ROS levels at the calvarial defect site were sufficient to trigger BMP-2 release from PTK-BAA films. To further confirm and quantify localized in vivo BMP-2 delivery kinetics from the responsive PEM films, lead-candidate PTK-BAA coated implants fabricated with 1.8 kDa PAA and fluorescently-tagged growth factor were directly compared against analogous PBAE-coated samples in the rat calvarial defect model. As visualized in Figure 8A, IVIS imaging of the Cy7-labeled protein indicated a decreased bolus release of BMP-2 in the responsive film coating.
Figure 6. Varying the PAA polyanion molecular weight (450, 5, or 1.8 kDa) in BMP-2/PTK-BAA films A) does not appreciably change the total BMP-2 loading as visualized by fluorescence imaging of films with Cy5-labeled protein. However, PAA1.8 samples B) display significantly more sensitivity to oxidation via BMP-2 release at both 1 and 0.1 mm H_2O_2 compared against higher molecular weight PAA formulations (n = 3 per treatment; *p < 0.05, **p < 0.005, ***p < 0.0005 for 1 mm H_2O_2 vs PBS; #p < 0.05 for 0.1 mm H_2O_2 vs PBS). All samples featured 30 tetralayer films constructed on PLGA scaffolds.

Figure 7. In vivo bone regeneration in critically-sized rat calvarial defects from oxidation-sensitive, BMP-2-coated implants. BMP-2/PTK-BAA LbL films (30 tetralayers) with varying PAA polyanion molecular weights (450, 5, or 1.8 kDa) were constructed on PLGA scaffolds and implanted into 8 mm diameter rat calvarial bone defects for 4 weeks. A) Non-invasive microcomputed tomography images of the calvarial bone were collected at week 4 post-surgery (original defect margins in red, new bone growth tinted blue, scale bar 5 mm, n = 4 animals per group). Quantitation of B) bone volume and C) tissue mineral density inside the defect margins demonstrated that the PAA1.8 films facilitated significantly more bone growth and mineralization than higher molecular weight PAA films (*p < 0.05), corresponding to the PAA1.8 samples’ increased BMP-2 release rates following oxidation.

Figure 8. In vivo BMP-2 release from ROS-sensitive PTK-BAA and conventional PBAE-coated implants in rat calvarial defects. As quantified from A) fluorescence imaging of Cy7-labeled BMP-2, growth factor delivery from the PTK-coated scaffolds B) exhibits a lower initial bolus discharge and is significantly extended compared against PBAE films. Importantly though, over 95% of the PTK film’s drug payload is released over a three-week time course. C) Quantification of half-life values for the respective release profiles (displayed as a 95% confidence interval) further demonstrates a nearly threefold increase in drug retention for PTK-BAA coatings. N = 3 animals per treatment, *p < 0.05.
formulations and significantly enhanced drug retention over three weeks in vivo (Figure 8B). Moreover, the calculated half-life values for the respective release curves indicated a nearly threefold increase in BMP-2 retention as quantified in Figure 8C. Given prior reports of bolus growth factor release from hydrolytically-sensitive PBAE films and the in vitro release kinetics demonstrated in Figure 4A, the significantly prolonged in vivo BMP-2 release kinetics from PTK-BAA coated implants is well-supported.

Crucially, this extended protein release did not appear to encourage indefinite BMP-2 retention in the PTK-BAA films as all samples discharged ≈80% of their BMP-2 by day 14 and 95% of their payload by day 21 (Figure 8B). These findings indicate that local ROS concentrations, quantified at less than 0.1 mm in most tissues, were sufficient to continuously degrade the oxidation-sensitive coatings and promote sustained payload delivery. However, it should be noted that the PTK-BAA/BMP-2 coatings appeared to elicit less bone regeneration than previously reported PBAE/BMP-2 LbL film formulations over the same 4-week time periods. Importantly, the PTK-BAA polymer and its degradation products did not cause significant toxicity in vitro (Figure S2, Supporting Information) or elicit an excessive local inflammatory response or necrosis in vivo as visualized in high-magnification hematoxylin and eosin-stained histology images (Figure S6, Supporting Information). Moreover, specific immunohistochemical staining for CD68-expressing macrophages in 4-week tissue sections did not indicate an abundant macrophage presence in these healing bone defects with blank or PTK-BAA coated implants (Figure S7, Supporting Information), further confirming the non-inflammatory nature of these degradable films.

Since toxicity or excess inflammation does not appear to cause the decreased calvarial bone regeneration achieved with these ROS-responsive films, the interplay between BMP-2 dose and release kinetics was additionally explored. Rapid BMP-2 release from a collagen substrate (as utilized in the commercial IN-FUSE system) requires BMP-2 doses between 1.25–2.5 µg to achieve full rat calvarial defect bridging and rapid delivery of nanogram-scale BMP-2 doses is insufficient to completely heal these rat bone injuries. However, previous work has also demonstrated negligible bone formation in rats when defects receive extremely slow rates of BMP-2 delivery. Conceivably, the strong electrostatic links in the PTK-BAA/BMP-2 assemblies which limited early film decomposition and prevented a corresponding initial bolus of discharged protein both in vitro and in vivo (Figures 6B and 8B) could have similarly hindered bone regeneration with these responsive films. Importantly, reports of robust rat bone formation at relatively low BMP-2 doses have indicated that growth factor release profiles featuring some initial release coupled with sustained BMP-2 delivery achieved superior bone regeneration outcomes.

To evaluate the hypothesis that a burst release of BMP-2 followed by more sustained growth factor delivery would improve bone healing, PLGA scaffolds were coated with “stacked” LbL films (constructed with lead-candidate 1.8 kDa PAA) featuring an inner 15 tetralayers of PTK-BAA/BMP-2 and an outer 15 tetralayers of PBAE/BMP-2 (Figure S8A, Supporting Information) and implanted in critically-sized rat calvarial defects for 4 weeks (Figure S8B, Supporting Information). Mediated by the PBAE components in the outer portion of the PEM assemblies, these combination film samples expectedly elicited a faster initial burst release of BMP-2 than films solely constructed with PTK-BAA polymers as confirmed by in vivo fluorescent tracking data in Figure S8C, Supporting Information. Interestingly, bone growth with these combination films was substantially greater in some of the treated animals but did not elicit a statistically significant increase in bone volumes across the entire treatment cohort when compared to animals treated with the single PTK-BAA film formulations (Figure S8D, Supporting Information). Though these findings do not exclude the possibility of some other PTK-BAA film variable inhibiting bone growth, they potentially give some indication of bone healing improvements that can be achieved by increasing initial in vivo BMP-2 delivery rates from these responsive systems.

3. Conclusions

ROS, ubiquitous biological signals in both healthy and diseased tissues, represent a promising cell-specific stimulus for promoting therapeutic delivery from PEM-coated implants. Here, newly synthesized PTK-BAA polycations were selectively degraded via oxidation and built robust multilayered PEM films upon LbL assembly with appropriate polyanions. When constructed with the therapeutic protein BMP-2, the PTK-BAA films produced ROS dose-dependent delivery of the bioactive growth factor while promoting in vitro cellular osteogenesis at levels corresponding with BMP-2 release kinetics. Moreover, these films’ in vitro sensitivity to oxidation could be further modulated by altering the polyanion used in LbL assembly to promote significantly increased rates of ROS-mediated protein liberation. This connection between tunable LbL components and enhanced BMP-2 delivery was further exhibited in critically-sized rat calvarial defects as implants coated in the most sensitive oxidation-responsive PEM films correspondingly generated a greater than 50% increase in new bone formation compared against less responsive film formulations. Moreover, this responsive BMP-2 release was evaluated against conventional hydrolytically-sensitive PEM coatings in vivo and demonstrated a nearly threefold extension in delivery half-life.

In expanding the future application space for these environmentally-responsive drug coatings, their ability to sustainably deliver therapeutics upon oxidative triggering is potentially useful for a number of indications. Inflammation-triggered delivery of anti-inflammatory drugs to improve tissue repair, ROS-stimulated release of chemotherapeutics to target cancer cells with high levels of oxidative stress, or immune-activated antibiotic discharge to help fight bacterial infections all represent promising future areas of exploration with these intelligent delivery systems. For continued development as a bone regenerative technology, the findings presented here further motivate the prospective development of ROS-responsive coatings with enhanced sensitivity that can more rapidly deliver payload upon local cellular activation to improve performance. In all, these collective data highlight the utility of PTK-BAA polymers as environmentally-responsive PEM film constituents and highlight the potential of tunable drug delivery systems for regenerating large-scale bone defects.
4. Experimental Section

**Materials**: All materials and reagents were obtained from Millipore-Sigma (Burlington, MA) or Alfa Aesar (Ward Hill, MA) unless otherwise specified. Silicon wafers were bought from Silicon Quest International (Santa Clara, CA). Linear polyethyleneimine (LPEI, 25 kDa) was purchased from Polysciences (Warrington, PA). Cy5 and Cy7 NHS ester dyes were procured from Lumiprobe (Rocky Hill, NJ). The alkaline phosphatase assay kit was bought from Abcam (Cambridge, United Kingdom). Recombinant human bone morphogenetic protein-2 (BMP-2) was generously donated by Bioventus LLC (Durham, NC) while the Human/Murine Interleukin-1α was obtained from BioLegend. A mouse monoclonal anti-BMP-2 antibody was obtained from R&D Systems. Recombinant human bone morphogenetic protein-2 was generously donated by Bioventus LLC (Durham, NC) while the Human/Murine Interleukin-1α was obtained from BioLegend. MC3T3-E1 mouse pre-osteoblasts were obtained from American Type Culture Collection (Manassas, VA) and cultured in Gibco Alpha Minimum Essential Medium (containing ribonucleosides, deoxyribonucleosides, 2 mM L-glutamine, and 1 mM sodium pyruvate, but without ascorbic acid) with 10% fetal bovine serum (Invitrogen) and 1% penicillin/streptomycin (Lonza). Cell Titer Glo assay kit was acquired from Promega (Madison, WI).

**Polyion Synthesis and Characterization**: Detailed experimental procedures for polymer synthesis and chemical characterization are provided in the Supporting Information. In brief, a thiolated diacrylamide monomer was synthesized using methods adapted from previous reports and polymerized by Michael addition with 4,4′-trimethylene diisocyanate for 24 h at room temperature in a 3:1 v/v solvent mix of ethanol/water, and then purified by 3x precipitation into cold THF. The generated PTK-BAA polymer structure was confirmed by 1H NMR, and molecular weight was determined using MALDI-TOF using a 2.5-dihydroxybenzoic acid matrix. The PBAE polymer was synthesized using previously established protocols and characterized by 1H NMR and gel permeation chromatography (GPC, Waters system with Styragnel Columns, THF mobile phase, linear polystyrene molecular weight standards). To synthesize the hexane-PBAE, a hexane diacrylamide monomer was generated from hexadimethylenetriamine and then polymerized with 4,4'-trimethylene diisocyanate as described above.

**Quantification of ROS-Triggered Thioketal Degradation**: PTK-BAA polymers were dissolved at 1 mg mL$^{-1}$ in deionized water and doped with 0, 0.1, 1, and 10 mM doses of H$_2$O$_2$. Samples were incubated for 72 h at 37°C before being lyophilized and then reconstituted in deuterated water to analyze with $^1$H NMR. The characteristic thiolactic peak (δ 1.59) was then normalized against the internal control peaks from the polymer’s diisocyanate units (δ 1.2–1.3). MALDI-TOF analysis of the 10 mM H$_2$O$_2$-incubated polymer was also used to determine degradation product molecular weight.

In Vitro Polymer Cytotoxicity: MC3T3-E1 mouse pre-osteoblast cells were seeded in a black-walled 96-well plate (5000 cells per well) for 24 h before treating with 100 µL of fresh media or media doped with 10 µg mL$^{-1}$ of 25 kDa LPEI, PBAE, PTK-BAA, or 10 mM H$_2$O$_2$-degraded PTK-BAA polymers (n = 3 to 6 biological replicates for each experiment). After 24 h of incubation, the plate was equilibrated to room temperature for 30 min before adding 50 µL of room-temperature Cell Titer Glo reagent to each well, shaking for 10 min, and then measuring luminescent signal using a Tecan Infinite 200 PRO microplate reader. All viable cell signals were normalized to no treatment wells.

**LbL Film Construction and In Vitro Drug Release with the Model Protein Lysozyme**: Complete procedures for protein labeling, film construction, and drug release quantification can be found in the Supporting Information. Briefly, polyelectrolyte solutions were formulated in 100 mM sodium acetate buffer (pH 5.0), with PTK-BAA at 0.5 mg mL$^{-1}$ and the 450 kDa PAA and fluorescein isothiocyanate (FITC)-labeled lysozyme (10% labeled protein by mass) being dissolved at 1 mM. Silicon substrates were plasma treated and then dipped into the polyelectrolyte bath using a Carl Zeiss HTM Series Programmable Slide Stainer, alternatively immersing in PTK-BAA (5 min), PAA (5 min), lysozyme (5 min), and PAA (5 min) with two 30 s rinses in deionized water between dips. Fabricated films were incubated in PBS, PBS with 0.1 mM H$_2$O$_2$, or PBS with 1 mM H$_2$O$_2$ before collecting releaseate samples over time to quantify fluorescence from the labeled lysozyme.

**PLGA Scaffold Fabrication**: Poly(lactic-co-glycolic) acid (PLGA: acid terminated, lactide/glycolide 50:50, molecular weight 38–54 kDa) was formed into a degradable scaffold using a diffusion-induced phase-separation protocol adapted from previously described methods. Raw polymer was fully dissolved at 25 w/w% in dimethylformamide (DMF), and the solution was deposited on a glass slide. Using a film-casting doctor blade dragged across the dissolved PLGA, a 1 mm-thick polymer solution was cast on the glass before being immediately submerged in a large excess of deionized water to induce phase separation. The rinse water was changed multiple times over 24 h before drying the membrane under vacuum.

**LbL Film Construction and Characterization with BMP-2**: Complete procedures for protein labeling, film construction, and PEM characterization can be found in the Supporting Information. LbL film construction with BMP-2 was carried out similarly to the methods described above for lysozyme samples. All polyelectrolyte solutions were prepared in 100 mM sodium acetate buffer, cationic polymers (PTK-BAA, PBAE, or hexane-PBAE) were prepared at 0.5 mg mL$^{-1}$ (pH 5.0), PAA polyanions (450 kDa, 5, or 1.8 kDa) were dissolved at 1 mg mL$^{-1}$ (pH 5.0), and BMP-2 (5% Cy5-labeled protein by mass for drug loading quantification samples, 40% Cy7-labeled protein for in vivo release samples) was formulated at 0.04 mg mL$^{-1}$ (pH 4.1). Plasma-treated substrates were immersed in the initial cationic polymer solution for 30 min before initiating the dipping program: PTK-BAA (5 min), PAA (5 min), BMP-2 (10 min), and PAA (5 min) with two 30 s rinses in deionized water between dips. This dipping protocol allowed for the coating of both the top and bottom scaffold surfaces. For PTK-BAA/Cy5-BMP-2/450 kDa PAA film samples with increasing numbers of tetralayer depositions, PEM thickness on silicon was quantified using a Dektak Stylus profilometer (Veeco Instruments Inc., Plainville, NY) while BMP-2 loading was quantified by fluorescence imaging using an IVIS Spectrum (PerkinElmer, Waltham, MA) and ELISA.

**In Vitro BMP-2 Release from LbL Films**: Additional experimental details can be found in the Supporting Information. Briefly, LbL films comprising BMP-2, 450 kDa PAA, and either PTK-BAA or PBAE degradable polycations were deposited on both top and bottom surfaces of PLGA scaffolds with 30 tetralayer repeats, incubated at 37°C in either PBS or PBS with 1 mM H$_2$O$_2$, and quantified for BMP-2 release using ELISA. For measuring responsive BMP-2 release from these same PTK-BAA films, coated PLGA samples were incubated at 37°C in either a continuous dose of 1 mM H$_2$O$_2$ or alternating between PBS and 1 mM H$_2$O$_2$ every 4 days. For both the “constant” and “pulsed” ROS-treated samples, releaseate was collected every 2 days and then quantified for BMP-2 amounts by ELISA. In determining the effects of PAA molecular weight on PEM protein, 30 tetralayer PTK-BAA/Cy5-BMP-2 films were fabricated with either 450, 5, or 1.8 kDa PAA on PLGA substrates, fluorescently imaged by IVIS to visualize and calibrate for in vitro BMP-2 loading, and finally incubated in PBS, PBS with 0.1 mM H$_2$O$_2$, or PBS with 1 mM H$_2$O$_2$ before quantifying BMP-2 release using ELISA. Control 30 tetralayer PEM films were also constructed with Cy5-BMP-2, 1.8 kDa PAA, and either the hydrolytically-sensitive PBAE or non-degradable hexane-PBAE polycations on PLGA substrates. Release experiments and quantifications were performed with the same methodology as used for PTK-BAA samples.

**In Vitro Osteogenic Bioactivity of Film-Released BMP-2**: Additional experimental details can be found in the Supporting Information. Briefly, releaseate from PTK-BAA/BMP-2/450 kDa PAA films incubated in PBS or 1 mM H$_2$O$_2$ was pooled across discrete time periods, titrated to equal volumes, and then lyophilized before re-suspending in culture media. MC3T3-E1 pre-osteoblasts were treated with the releaseate-doped media for 48 h with an additional 4-day incubation in osteogenic differentiation media before quantifying alkaline phosphatase activity as normalized to total cell protein content.

**In Vivo Rat Calvarial Defect Procedures**: All animal work was approved by the Committee on Animal Care at the Massachusetts Institute of Technology (protocol 0718-057-21, PI Dr. Paula Hammond), and adult male Sprague-Dawley rats (300–350 g, Taconic) were used in all procedures. Critically-sized calvarial defect studies were carried out using well-established methods. First, anesthetized animals were cranially shaved, disinfected using betadine and alcohol, and given pre-operative...
meloxicam (anti-inflammatory) and buprenorphine sustained release (analgesic). Next, an incision through the scalp was used to expose the soft tissue/calvarium before using blunt dissection to displace the periosteum covering the bony surface. With the bone exposed, an 8.0 mm-diameter dental trephine drill with intermittent sterile saline irrigation was used to create a circular defect in the calvarial bone. Without disturbing the dura or superior sagittal sinus, the calvarium was excised before press-fitting the respective PEM-coated PLGA scaffold sample (also 8.0 mm-diameter) into the surgical bone defect. Finally, the periosteum was closed over the bone and implant using 5-0 monofilament absorbable sutures before also suturing the skin over the periosteum along the initial incision. Animals were given post-operative anti-inflammatory and analgesics until recovery. N = 4 animal cohort sizes were chosen per treatment group based on power analyses from previous reports with a calvarial defect model.[22a] In vivo studies employing unlabeled BMP-2 treatment group based on power analyses from previous reports with a calvarial defect model.[22b] In vivo release of fluorescently-tagged BMP-2 (Figure 8 and Figure S8C, Supporting Information) ultimately only featured n = 3 animals per treatment group due to humane euthanasia of animal subjects following surgical complications from excessive blood loss. Microcomputed tomography, IVIS fluorescence imaging, and histological sample preparation details are described in the Supporting Information. Whole slide imaging analysis and visualization were performed using QuPath software.[54]

Statistical Analysis: All data are presented as the mean and standard deviation unless otherwise indicated. Statistical analyses were performed using Student’s t-test for single comparisons or single factor analysis of variance (ANOVA) with Tukey post-hoc tests for multiple comparisons. p-Values less than 0.05 were considered statistically significant.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflict of Interest

P.T.H. discloses her competing interests as co-founder of LayerBio Inc. and service on the advisory boards of Performance Indicator and Moderna Therapeutics. All other authors declare no conflict of interest.

Data Availability Statement

Research data are not shared.

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BMP-2, bones, layer-by-layer, reactive oxygen species, responsive drug delivery
[20] a) B. D. Almqist, S. A. Castleberry, J. B. Sun, A. Y. Lu, P. T. Hammond, Adv. Healthcare Mater. 2015, 4, 2090; b) M. Bouyer, R. Guil-lot, J. Lavaud, C. Plettinx, C. Olivier, V. Curry, J. Boutonnat, J.-L. Coll, F. Peyrin, V. Josserand, G. Bettega, C. Picart, Biomacromolecules 2016, 104, 168.

[21] a) Y. Min, P. T. Hammond, Chem. Mater. 2011, 23, 5349; b) J. Min, R. D. Braatz, P. T. Hammond, Biomaterials 2014, 35, 2507; c) J. Min, K. Y. Choi, E. C. R. Padera, R. D. Braatz, M. Spector, P. T. Hammond, ACS Nano 2016, 10, 4441; d) K. C. Wood, H. F. Chuang, R. D. Batten, D. M. Lynn, P. T. Hammond, Proc. Natl. Acad. Sci. USA 2006, 103, 10207.

[22] a) N. J. Shah, M. N. Wyder, M. A. Quadir, N.-M. Dorval Courchesne, H. J. Seehrer, M. Nevins, M. Spector, P. T. Hammond, Proc. Natl. Acad. Sci. USA 2014, 111, 12847; b) N. J. Shah, M. L. Macdonald, Y. M. Beben, R. F. Padera, R. E. Samuel, P. T. Hammond, Biomaterials 2011, 32, 6183.

[23] N. J. Shah, M. N. Wyder, J. S. Moskowitz, M. A. Quadir, S. W. Morton, H. J. Seehrer, R. F. Padera, M. Spector, P. T. Hammond, Sci. Transl. Med. 2013, 5, 191ra83.

[24] X. Guo, M. C. Carter, V. Appadoo, D. M. Lynn, Biomacromolecules 2019, 20, 3464.

[25] K. Hensley, K. A. Robinson, S. P. Gabbita, S. Salsman, R. A. Floyd, X. Guo, M. C. Carter, V. Appadoo, D. M. Lynn, Antioxid. Redox Signaling 2014, 20, 1126.

[26] a) L. Key, W. Wolf, C. Gundberg, W. Ries, Bone 1994, 15, 431; b) I. R. Garrett, B. F. Boyce, R. Oreffo, L. Bonewald, J. Poser, G. R. Mundy, J. Clin. Invest. 1990, 85, 632.

[27] a) J. R. Martin, C. L. Duvall, in Oxidative Stress and Biomaterials, Vol. 1 (Eds.: T. Dziubla, D. A. butterfield), Academic Press, Cambridge, MA 2016, p. 225; b) F. El-Mohtadi, R. d’Arcy, J. B. Sun, A. Khademhosseini, Adv. Healthcare Mater. 2020, 9, 1901714.

[28] X. Guo, M. C. Carter, V. Appadoo, D. M. Lynn, Biomacromolecules 2019, 20, 3464.

[29] K. Hensley, K. A. Robinson, S. P. Gabbita, S. Salsman, R. A. Floyd, Free Radicals Biol. Med. 2000, 28, 1456.

[30] P. Pacher, J. S. Beckman, L. Liaudet, Physiol. Rev. 2007, 87, 315.

[31] M. Mittal, M. R. Siddiqui, K. Tran, P. S. Reddy, A. B. Malik, Antioxid. Redox Signaling 2014, 20, 1126.

[32] a) J. R. Martin, C. L. Duvall, in Oxidative Stress and Biomaterials, Vol. 1 (Eds.: T. Dziubla, D. A. Butterfield), Academic Press, Cambridge, MA 2016, p. 225; b) F. El-Mohtadi, R. d’Arcy, J. B. Sun, A. Khademhosseini, Adv. Healthcare Mater. 2020, 9, 1901714.

[33] a) J. R. Martin, C. L. Duvall, in Oxidative Stress and Biomaterials, Vol. 1 (Eds.: T. Dziubla, D. A. Butterfield), Academic Press, Cambridge, MA 2016, p. 225; b) F. El-Mohtadi, R. d’Arcy, J. B. Sun, A. Khademhosseini, Adv. Healthcare Mater. 2020, 9, 1901714.