Research Article

Manufacturing of Bathroom Wall Tile Composites from Recycled Low-Density Polyethylene Reinforced with Pineapple Leaf Fiber

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1. Introduction

Plastic materials are used widely for packaging drinks and household goods because of its versatility. However, due to its nonbiodegradable nature, it is affecting the environment through pollution. For this reason, there is an urgent need to remove the plastic waste from the environment. These plastics can be removed by burning, reusing, or recycling. Burning of the plastic waste produces hazardous fumes which are toxic, whereas reusing it is not an attractive alternative because of contamination. For this reasons, recycling to other products turns out to be a more attractive choice [1]. Waste plastics can be chemically recycled in many ways such as glycolysis [2], hydrolysis, alkalosis, methanolysis, ammonolysis, and aminolysis [3, 4]. A very attractive way of removing waste plastics is by mechanical recycling, which consist of gathering, shredding, and pelletizing followed by their reintroduction into the manufacturing of other plastic products. The recycled plastics can also be reinforced with either natural or synthetic textile fibers to manufacture load-bearing composite materials.

The interest in natural fiber-reinforced composite (NFRC) is increasing because of their properties of biodegradability, noncarcinogenicity, cost effectiveness, ecofriendliness, absence of health hazards, easy collection, and regional availability. They are also a renewable resource, thus providing a better solution for supply sustainability [5]. The NFRC versatile characteristics make it suitable for automobiles, railway coach, building construction, wall partition, cabinets, furniture, and packaging manufacture [6]. NFRC are viable because of the wide availability of natural fibers and agriculture by-product fibers which can be used as reinforcements [7]. Researchers have been working round the clock to find new sources of natural fibers that possess comparable physical and mechanical properties to synthetic fibers to be used in reinforcing of composites [8]. This is because synthetic fiber-reinforced polymers are costly and have an impact on the environment. Despite the advantages, natural fibers have
higher moisture absorption and they are hydrophilic in nature thus hampering composite fabrication. Moisture absorption swells and softens the fibers, thus adversely reducing their mechanical properties while its hydrophilic nature affects its dispersion/mixing within the matrix phase. For this reasons, natural fibers needs to be modified either physically or chemically to increase the interaction between the fiber and the polymer matrix. A number of studies has been done to solve this phenomenon [9, 10]. Ali et al. [11] studied the effect of fluorocarbon and hydrocarbon on the mechanical properties and moisture regain of jute fiber-reinforced composite materials. They observe that the composites made with treated jute fiber regained very low moisture content as well as showed improved mechanical properties. Alix et al. [12] modified flax fibers with silane (Si) and styrene (S) treatments and studied the water behaviour of their polyester composites. They observe that treatment increased the composites’ moisture resistance. Bakri et al. [13] studied the effects of alkaline treatment on the mechanical, morphological, and spectral properties of banana fiber/epoxy composites and noted that treated fiber composites had superior properties.

There are a number of works done on natural fiber-reinforced composites using fibers such as kenaf, oil palm, bamboo, jute, sisal, coconut, and pineapple leaf fibers as shown in Figure 1 [14]. Pineapple leaves are waste agricultural materials and can be used as a source of fibers [15]. Both the thermosets and thermoplastic resins have been used as matrices with these natural fibers. As a thermoplastic matrix material, low-density polyethylene (LDPE) has been extensively used with natural fibers for composite manufacture.

The development, mechanical properties, and uses of pineapple leaf fiber- (PALF-) reinforced polymer composites have been reviewed [15, 16]. George et al. studied the mechanical properties of short PALF-reinforced LDPE composites prepared by melt mixing and solution mixing [17]. They evaluated the influence of fiber length, fiber proportion, and fiber orientation. Besides, fiber breakage and damage during processing were analysed using fiber distribution curve plus optical and scanning electron micrographs. PALF-epoxy composites have also been study by Jain et al. [18]. They prepared composites using the hand layup method and studied the effect of fiber proportion on the morphological, chemical, mechanical, and thermal properties of the composites. PALF have also been used to prepare polypropylene-based composites by twin-screw extrusion and deviation in fiber length, diameter, and aspect ratio induced by the mixing process analysed by Berzin et al. [19].

This research is aimed at using manually extracted PALF to reinforce recycled LDPE. The manufactured composites were evaluated for their tensile strength, flexural strength, impact resistance, and water absorption properties.

2. Materials and Methods

2.1. Materials. Recycled LDPE was sourced from Addis Ababa, while urea, sodium hydroxide, and acetic acid were sourced from the market and used without alteration. Pineapple leaves were collected from the southern parts of Ethiopia.

2.2. Extraction of Pineapple Leaf Fiber. The fibers were extracted from pineapple leaves using combined aqueous and mechanical methods. The leaves were scratched using a blunt blade to remove the waxy layer from its surface and immersed in urea solution for one week at room temperature. The urea solution was prepared by dissolving 10 g/l of urea. Urea is an organic complex that is extremely soluble in water and nonpoisonous. It is frequently utilised as an enricher for the majority of the crops. It also encourages the development of microorganisms in soil and water. Urea escalates the wetting action of water and heightens the growth of microbes in water. Therefore, urea was used for fiber extraction in order to quicken the retting process [20, 21]. The leaves were removed and the fibers separated using a blunt knife by eliminating the leaf covers still attached to the fiber surface. The extracted fibers were washed with water and dried in air for 24 hours.
2.3. Chemical Treatment of Pineapple Leaf Fiber. Prior to composite manufacture, the fibers were treated with sodium hydroxide solution in order to improve the fiber-matrix interface. The fibers were treated with 5% NaOH solution for one hour at room temperature, after which they were washed several times and neutralized with acetic acid [22]. The fibers were dried in an air blast oven at a temperature of 60°C and stored in a dry environment ready for composite manufacture.

2.4. Fiber Testing. Tensile testing of extracted fibers were carried out in an Instron Universal Testing Machine Model 1121 at a crosshead speed of 1 mm per min. Specimens were prepared by mounting single fibers on a stiff cardboard piece with a 50 mm window. The ends of the fibers were fixed on the cardboard. The diameters of the extracted fibers were measured using an optical Leica microscope. Since the diameter of the fibers varies at different sections, an average of ten readings was taken for diameter determination. Moisture content of the fiber was also evaluated according to the ASTM D 1909–13 test standard.

2.5. PALF-LDPE Composite Preparation. Randomly oriented PALF-LDPE composites with varying fiber length and fiber weight proportion were manufactured by the melt-mixing process. The parameters used were a mixing time of 8 min, rotor speed of 60 rpm, and mixing temperature of 130°C based on an early work by George et al. [17]. The temperature of 130°C was used because it does not affect the fiber properties. Composite tiles of size 200 × 150 × 8 mm were prepared using a closed mold. The mold was polished with a release agent to avoid LDPE from sticking to it. The process involved melting of the shredded LDPE, adding predetermined proportion of the chopped fibers, melt-mixing thoroughly to form a homogeneous viscous solution, and placing it into the prepared mold. Finally, the mold was closed and the samples were cooled down to room temperature under 12.5 MPa pressure for 30 min. The specimens of the tiles produced were shaped by sandpaper and used for testing as shown in Figure 2.

2.6. Composite Tests

2.6.1. Tensile Test. Tensile tests were performed on Instron 5567 at a cross-head speed of 50 mm/min. The samples were prepared for this test according to the ASTM D638 test standard (type II). At the beginning of the tensile test, the specimen elongates and the resistance of the specimen increases which was detected using a load cell. This value was recorded until the specimen fractured, and five samples were tested.

2.6.2. Flexural Test. Flexural strength is the combination of tensile strength and compressive strength. The tests were done on a universal testing machine. The specimens were prepared according to the ASTM D790 test standard with dimension 200 × 30 × 8 mm. The specimens were tested on a support span of 130 mm as per the standard.

2.6.3. Impact Test. Charpy impact tests on unnotched specimens were performed using a pendulum impact testing machine JBS-300 N model. The test specimens were prepared according to the ASTM D6110-18 test standard with a dimension of 50 mm long, and a cross-sectional area of 24 mm². Five specimens were tested and an average value was reported.

Table 1: Properties of extracted pineapple leaf fibers.

| Properties                  | Values  |
|-----------------------------|---------|
| Average length (mm)         | 300.5   |
| Average diameter (μm)       | 59.73   |
| Moisture content (%)        | 12      |
| Tensile strength (cN/Tex)   | 32.679  |
| Elongation at break (%)     | 2.08    |
2.6.4. Water Absorption Test. Water absorption tests were carried out in accordance with the ASTM D570 test standard. Samples of each composite type were oven dried before its weight was recorded as the initial weight of the composites. The samples were then placed in distilled water maintained at room temperature (25°C) for 24 hours. The samples were then removed from water, dried with a cotton fabric, and weighed. The amount of water absorbed by the composites (in percentage) was calculated using equation (1):

\[
\%W = \left(\frac{W_t - W_o}{W_o}\right) \times 100
\]

where \(W_t\) is the weight of the composite after immersion in water and \(W_o\) is the weight of a dried sample.

3. Results and Discussion

3.1. Characterization of PALF. The physical and mechanical properties of the extracted PALF are shown in Table 1.

3.2. Characterization of the Manufactured Composites. From the experiments, the mechanical and water absorption properties of the manufactured composite are shown in Table 2. By using the design of experiment software, the properties of the fiber proportions not tested were extrapolated as also reported in Table 2.

3.3. Tensile Strength

3.3.1. Effect of Fiber Weight Proportion on Tensile Strength. As seen in Figure 3, the tensile strength of the composite increased with the increase in fiber weight (weight.) proportion from 10% to 30% and from 800 N to 1200 N, respectively, but after 30% fiber proportion, the strength decreased. In other words, the addition of 20 weight. % of PALF increased the strength of the composite by almost 50%. From observation, the optimum fiber proportion which yields the highest tensile strength was at 30%. This increase of strength means that the reinforced LDPE became stiffer and could withstand higher load. The fiber served as reinforcement because the major share of load was taken up by the fibers [23]. Besides, at fiber weight, % greater than 30%, the fibers were excessive; therefore, the matrix LDPE was not enough to flow through and wet each and every fiber thus leaving voids and the fibers were easily exposed to environmental degradation. The interfacial adhesion between fiber and LDPE was not good at these levels of weight proportion as shown in Figure 4. Fiber agglomerations occur thus causing fiber dispersion problems in LDPE, which led to a decrease in tensile strength [17].

3.3.2. Effect of Fiber Length on Tensile Strength. The strength of fiber-reinforced composites depends on the degree to which an applied load is transferred by the matrix to the reinforcing fibers. The extent of load transfer is a function of fiber...
length and magnitude of fiber-matrix interfacial bond. In short-fiber-reinforced composites, there exists a critical fiber length that is required for the fiber to develop its fully stressed condition in the matrix. If the fiber is shorter than this critical length, the stressed fiber will be pulled out from the matrix and the composite will fail at a lower stress. When the length is greater than the critical length, the stressed composites will lead to breaking of fibers and a higher composite strength. From Figure 5, it can be seen that as the fiber length is greater than the critical length, the stressed composite will fail at a lower stress. When this critical length, the stressed fiber will be pulled out from the matrix and the composite will fail at a lower stress.

3.3.3. Effect of Fiber Weight Proportion as a Function of Fiber Length on Tensile Strength. As seen in Figure 6, the tensile strength of the PALF-reinforced LDPE composite increased with an increase in both fiber weight proportion and fiber length up to 30% and 30 mm, respectively. At this point, the optimal tensile strength was achieved, but above this value any increase in one factor or in both resulted in a reduction in tensile strength due to the formation of the fiber-to-fiber interaction rather than the fiber-to-polymer interaction. In addition, if the fiber length is above 30 mm, there is a chance of fiber entanglement within the composite which can result in a significant reduction in its tensile strength.

3.4. Flexural Property

3.4.1. Effect of Fiber Weight Proportion on Flexural Strength. The flexural strength for PALF-LDPE composite is shown in Figure 7. From the figure, the flexural strength increased gradually with the increase in fiber weight proportion. An increase in fiber weight content from 10 to 30% on weight increased the flexural strength by about 42%. This could be due to more fibers being present on a given composite cross-section to carry the load at a higher fiber weight %. However, further increase in fiber weight content above this value resulted in the lowering of flexural strength. The decrease in flexural strength at higher fiber weight proportion may be due to the dispersion problems during melt-spinning [23].

3.4.2. Effect of Fiber Length on Flexural Strength. The reliance of composite flexural properties on fiber length is shown in Figure 8. The same trend of higher flexural strength is shown with fiber length of 30 mm when compared to the fibers which are shorter and also fibers which are above 30 mm composites. The flexural strength of the PALF composites containing 30 mm long fibers was 65% higher than that of PALF composites with 10 mm length fiber. The optimum flexural strength and modulus of the PALF composites were obtained at a fiber length of 30 mm. Similar results were reported by Devi et al. where the flexural strength of the composite containing 30 mm long fibers was 25% higher than that of composites containing 5 mm length [6].

3.4.3. Effect of Fiber Weight Proportion as a Function of Fiber Length on Flexural Strength. From Figure 9, the variation of flexural strength with fiber weight proportion and fiber length combined can be seen. From the figure, the flexural strength values of the PALF-LDPE composites were found to increase with both increments in fiber weight % and fiber length. The optimum flexural strength was achieved at 30% fiber weight proportion and 30 mm length of fiber. The increase in flexural strength was significantly affected by the increase in fiber weight % rather than the increase in fiber length as shown by the gradient of the increase.

3.5. Impact Strength

3.5.1. Effect of Fiber Weight Proportion on Impact Strength. Figure 10 depicts the Charpy impact strength (CIS) of unnotched samples of treated PALF composites with fiber weight proportion varying from 10 to 30%. CIS increased
with the amount of fibers added until a reduction was reached above 30% fiber weight. At this fiber weight %, the CIS value was about 20% more than that of the 10% fiber-reinforced LDPE as the fiber bridge cracks and increases the resistance of its propagation. However, at higher percentage of fiber weight % above the optimal percentage (30 weight%), the CIS decreased than the 30 weight% since addition of more fibers creates regions of stress concentrations that require comparatively less energy to initiate a crack as seen in Figure 4.

Piah et al. [24] reported that the energy-absorbing mechanism of composites during fracture includes the utilisation of energy required to debond the fibers and pull them completely out of the matrix due to weak interface strength between the fiber and matrix. In practical interest, a significant part of energy absorption during impact takes place through the fiber pull-out, matrix crack, and fiber breakage.

### 3.5.2. Effect of Fiber Weight Proportion as a Function of Fiber Length on Impact Strength

The work of fracture (impact strength) of PALF-LDPE composites at 30% proportion as a function of fiber length is shown in Figure 11. It is seen that comparatively higher impact strength is observed for composites of fiber length 30 mm (critical fiber length). In fact, there was a decrease in impact strength for composites of higher fiber length (i.e., >30 mm).

In most fiber-filled composites, a significant part of the energy absorption during impact takes place through the fiber pull-out process. The energy involved and, hence, toughness is higher when the length of the fibers is equal or greater to the critical length (lc). Fibers shorter than lc (30 mm) will be pulled out from the matrix rather than being broken when a crack passes through the composite. The fracture energy will then be basically a combination of the work necessary to debond the fibers out of the matrix and the work done against friction in pulling the fibers out of the matrix as shown in Figure 4. In a similar manner, when the fiber length is above the critical length (30 mm), the impact strength of the composite is decreased due to weak surface interaction. It requires less energy to overcome the fiber-fiber interaction.
rather than the fiber-polymer interaction; therefore, it exhibits lower impact strength. The weak surface adhesion between the fiber and the matrix initiated the crack up on the energy transferred to the composite. As reported by Chong et al. [25], impact strength decreases due to poor interfacial bonding between the fiber and the matrix.

3.6. Water Absorption

3.6.1. Effect of Fiber Weight Proportion as a Function of Fiber Length on Water Absorption. All lignocellulosic fibers have low resistance to water absorption due to the presence of OH groups in its chemical structure. After 24 h immersion in water, there was a noticeable effect of fiber content on water absorption test results as shown in Figure 12. The lowest water absorption rate was achieved with 10% fiber weight proportion. There was only a slight reduction in the rate of water absorption when the length of the fiber was increased from 10 mm to 30 mm. When the volume of the fiber increased from 10% to 30%, the rate of water absorption also increased because more lignocellulosic fibers are added into the composite, meaning that more hydrogen bonds were formed between the water molecules and OH group in the fibers. The same results were observed by Huner et al. who reported that the rate of water absorption increased with the increase in fiber content [26, 27]. This was due to the formation of less surface interaction between the matrix and fiber when mixed together giving higher water absorption [28].

4. Conclusion

The results of this study showed that a useful composite with good properties can be successfully manufactured by reinforcing waste LDPE with treated PALF. The optimum length of the fiber required to obtain PALF-LDPE composites of maximum properties was found to be 30 mm. The tensile strength of PALF-LDPE composites increased drastically up to the optimum level of fiber weight proportion. But, in the case of flexural strength, it increased linearly with the weight fraction and fiber length up to the optimal level of 30% fiber weight proportion and 30 mm fiber length, beyond which there was reduction in the flexural strength. The impact strength also increased linearly with the fiber weight proportion. The composite with 30 weight % fiber content exhibited an impact strength of 225.2 J/mm². The amount of water absorbed by the composites increases with the increase in the PALF weight proportion due to the formation of interaction between the OH groups in fibers and water. Therefore, the optimal mechanical and water absorption properties were achieved at 30% fiber proportion and 30 mm fiber length. The manufactured composites showed improved mechanical properties with respect to the fiber proportion increase, showing the reinforcement potential of PALF. It is thus possible to envision a prospective industrial use of this agricultural waste, for instance, for the manufacture of bathroom wall tiles.
Data Availability

All relevant data to the manuscript have been included.

Conflicts of Interest

The authors declare that they have no conflicts of interest.

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