Comparative Analysis Between Fly Ash Geopolymer and Reactive Ultra-Fine Fly Ash Geopolymer

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Abstract

This study investigates novel geopolymers by combining Reactive Ultra-fine Fly Ash (RUFA) with 4M sodium hydroxide as an alkali activator. Comparing with general fly ash geopolymers, RUFA geopolymer pastes are characterized in terms of compressive strength, microstructure, and crystalline phases. The RUFA geopolymer is successfully obtained as alumina-silicate bonding materials with the same properties as the general fly ash-based geopolymer. The high compressive strength of the RUFA-based geopolymer samples (13.33 MPa) can be attributed primarily to Ca-based alumino-silicate hydration products and Na-based alumino-silicate complexes. This research presents an innovative application for geopolymers using RUFA. In the follow-up study, the influence of synthesis and concentration of alkali activator can be considered in RUFA-based geopolymers.

Keywords: reactive ultra-fine fly ash, alkali activator, spherical particles, microscopic properties, amorphous

1. Introduction

1.1. Background and motivation

The explosion of the world’s population in the past few decades has created the problem of housing shortages. Most buildings are built by using cement, creating massive amounts of CO\textsubscript{2} (the leading culprit of the greenhouse effect) [1-3]. Greenhouse gases retain the energy by absorbing infrared radiation and thereby cause the gradual temperature rise at the earth’s surface, which accelerates global warming. To mitigate the greenhouse effect, governments should focus on devising effective measures to reduce greenhouse gas emissions. Cutting down on cement use would not only reduce CO\textsubscript{2} emissions and slow down the greenhouse effect, but also promote energy conservation. Portland cement is one of the primary materials used in construction. Approximately 3 billion tons of Portland cement is produced every year [4]. However, for every ton of Portland cement produced, 0.55 tons of CO\textsubscript{2} is created via chemical reactions and 0.39 tons of CO is generated by combustion and fuel emissions during the production process, which adds up to approximately 0.94 tons of CO\textsubscript{2} [5]. An investigation report in 2000 [6-7] revealed that an average of 0.87 kg of CO\textsubscript{2} is emitted for every kilogram of cement produced and that the cement
manufacturing industry contributes to 7% of global CO$_2$ emissions [8]. The report further forecasted that the global demand for Portland will double to 6 billion tons per year within the next 40 years [9]. Therefore, the use of suitable materials as an alternative to cement in construction materials is an important research issue.

1.2. Literature survey

Geopolymer is one kind of low carbon emission green composites. It is formed through the process of geopolymerization after alkaline solution reacts with such industrial by-products or wastes. Geopolymer is an inorganic alumino-silicate compound and it is usually produced from conventional pulverized coal fly ash. An emerging technology for stabilizing the frame-like polymerization structure in geopolymer can help with energy saving and carbon reduction [10]. This new generation of green materials may also replace ordinary cement and concrete in the future. Geopolymer is composed of SiO$_4$ and AlO$_4$ tetrahedrons sharing oxygen atoms to form a closed frame structure of Si-O-Al, which is similar to Zeolite. Its hardening mechanism is similar to that of conventional cement. It is the inorganic polymerization between the particles of the system colloid. The hardening time is about 1.5 to 2.0 hours at room temperature, which is one kind of early-strength material [10-11].

The hardening mechanism of geopolymer is different from conventional cement in that the cement is hardened as a hydration bond, while the geopolymer is a chemical bond. At this stage, the development of geopolymer into a new generation of environmentally friendly materials has considerable real potential. Geopolymer should be multi-functional materials and have better properties than traditional cement-based materials, including rheological, mechanical, thermal properties, process properties, durability, and so on [12-13]. It has been already proven that some materials properties of geopolymer composites are comparable or even better than those of the traditional cementitious materials, which are another positive finding in addition to the environmental aspect. Albitar et al. [14] concluded that geopolymer concrete are more chemically stable and superior to conventional concrete in an acidic environment, and exhibit lower deterioration of mechanical properties under chemical attacks.

An increasing burden on the environment caused by industrial activities (involving non-negligible contribution of cement production) creates a demand for seeking for reliable alternative non-cement construction materials possessing comparable properties with cementitious materials. There are various ways of preparation of promising non-cement materials which are based on partial replacement of cement by other eco-friendlier constituents. Non-cement materials produced by using various precursors and alkalis, or blending of materials, can react together and form a solid matrix of reasonable mechanical performance. The geopolymers made with waste materials, such as fly ash and ground granulated blast furnace slag, are attracting widespread attention as a green alternative to Portland cement [15-16]. These materials have also been widely adopted in the construction industry due to their excellent mechanical properties and durability [17-18]. Reactive Ultra-fine Fly Ash (RUFA) is an industrial by-product from thermal power plants heated to 1300°C, with the same chemical composition as conventional fly ash (heated from 900 to 1100°C) but smaller spherical particles ranging in size from 0.1 µm to 10 µm [19]. The large surface area provided by the superfine particles greatly enhances pozzolanic activity, sufficient for the partial replacement of Portland cement [19].

1.3. Purpose and organization

RUFA has spherical particles of an amorphous structure and is a new industrial by-product from thermal power plants. The chemical composition of RUFA is similar to conventional pulverized coal fly ash and it is suitable for manufacturing geopolymeric binder which can occur under curing conditions at elevated temperatures between 50 to 80°C as well as considered a cleaner process due to much lower CO$_2$ emission. In this study, we examine the feasibility of creating RUFA-based geopolymers using sodium hydroxide as an alkali activator. It also presents a comparison of geopolymers made
on the conventional fly ash and RUFA. Polymer specimens are characterized in terms of chemical composition, compressive strength, and microstructure. Fig. 1 shows the flow chart for the RUFA geopolymer research approach. The process consists of five steps:

1. To analyze the physical and chemical analysis of RUFA particles and traditional fly ash;
2. To formulate a suitable mix design of RUFA geopolymer;
3. To cast the specimens and conduct the testing preparation;
4. To use evaluation tests, including X-ray Diffraction (XRD) analysis, compressive strength test, Scanning Electron Microscopy (SEM) observations, Energy Dispersive Spectroscopy (EDS) analysis, and Thermogravimetric and Differential Thermal Analysis (TG-DTA);
5. To evaluate the feasibility of RUFA geopolymer.

![Creative RUFA geopolymer research methodology](image)

### 2. Test Methods

#### 2.1. Materials

The RUFA used in this study is obtained from a power plant in Taiwan (TRIAXIS Corporation). The light gray RUFA comprises spherical particles ranging in size from 0.1 µm to 5.0 µm, most of which are amorphous phases, as indicated by a broad hump in XRD patterns at 2θ between 20° and 35° [19]. The specific surface area of RUFA is 3380 m²/kg and the specific gravity is 2.67, respectively. The SEM photo and EDS result of the spherical particles of RUFA are shown in Fig. 2 and Table 1, respectively.

The composition of the RUFA is as follows: silicon dioxide (49.11 %), aluminum oxide (28.07 %), and calcium oxide (8.84 %) by the results using X-ray fluorescence. The primary oxide content in RUFA is similar to that of general class F fly ash (SiO₂+Al₂O₃+Fe₂O₃ ≥75 wt.%). At 28th days, the pozzolanic strength activity index of RUFA (107%) is higher than that of general fly ash (72-83%) [19]. Sodium hydroxide (NaOH) beads of 98 wt.% purity are mixed with deionized water at a concentration of 4.0 M for use as an alkaline activator.
2.2. Mix design, specimens, and test methods

The composition of the RUFA-based geopolymer pastes is RUFA and sodium hydroxide. RUFA geopolymer specimens are prepared by mixing RUFA with 4.0 M NaOH solution (160g of NaOH powders are added in 1000ml deionized water) at a constant liquid/binder ratio of 0.30 for 5 min in paste mixing process. The fresh pastes are immediately poured into cubic molds (50×50×50 mm) and shaken on a vibration table for 2 min to remove entrained air, and then held at 50°C for 24 h using the oven. After initial curing, the specimens are removed from the molds to continue curing at room temperature until testing.

Compressive strength tests are conducted at curing ages of 7th, 14th, and 28th days, in accordance with ASTM C109 standards. The compressive strengths in this study are averaged from five specimens. Following compression testing, the specimens are ground into powders for subsequent analysis via SEM, EDS, XRD and thermal analysis. SEM and EDS examinations are performed at 7th and 28th days, in accordance with ASTM C1723 standards. TG-DTA is conducted at a curing age of 28th days at temperatures ranging from 20°C to 1000°C under a heating rate was 10°C/min. XRD analysis is scanning at 2θ from range for 10° to 65° at the age of 28th days in accordance with ASTM C1365.

3. Results and Discussion

3.1. Feasibility of RUFA and traditional fly ash geopolymer in paste mixture

The liquid/binder ratio of traditional fly ash-based geopolymer mixed with the sodium hydroxide as alkaline activator is between 0.4 and 0.6. For a lower liquid/binder ratio of 0.3, a mixed activator of sodium hydroxide and sodium silicate is suitable as an alkali activator, but its workability will be considerably reduced. The liquid/binder ratio of traditional fly ash-based geopolymer is too low to make it mix into geopolymer pastes. At a liquid/binder ratio of 0.30 in RUFA geopolymer, the geopolymer pastes can be evenly mixed and the mix is very viscous when the mold is lifted away from the paste as shown in Fig. 3(a). As shown in Fig. 3(b), the geopolymer pastes are immediately dropped at the table 25 times in 15 sec in accordance with the standard of ASTM C1437. RUFA geopolymer pastes are performed better fluidity as the flowability is between 90 to
100%. The spherical RUFA particles have good water retention and cohesion as well as the increased lubricant effect. It is mentioned that the fluidity of RUFA geopolymer is improved [19]. When RUFA reacts with alkali activator forming geopolymers, its finer surface gets reactivated reaction and amorphous reaction products. Higher pozzolanic strength activity index and finer particles of RUFA [19] benefit the RUFA geopolymer pastes to be evenly mixed. This is verified that it is feasible for RUFA to mix NaOH as the geopolymer pastes.

3.2. Compressive strength and XRD patterns

As shown in Fig. 4, the compressive strength increases from 6.17 MPa at 7th days and 10.07 MPa at 14th days to 13.33 MPa at 28th days. Note that the strength at 28th days is close to that of geopolymer made using traditional fly ash with 4M NaOH as activator (15 MPa) [20-21]. The increase in strength is proportional to the quantity of alumino-silicates provided from the RUFA. RUFA contains large quantities of Al (28%) and Si (49%), which react with NaOH in geopolymer (general fly ash contains the quantities of Al (21%) and Si (48%)). We also infer trace quantities of calcium silicate hydrates (C-S-H) and sodium alumino silicate (N-A-S). As shown in Fig. 5, OH⁻ ions from the alkaline solution catalyze the reaction process, which promotes the dissolution of Si⁴⁺ and Al³⁺ ions from the RUFA. XRD peaks indicative of Tobermorite [CaSiO(HO)₂·4H₂O], Pezzottaite (CsAl₂Si₆O₁₈), and Hedenbergite (CaFeSi₂O₅) reveal the formation of sodium alumino-silicate hydrate (N-A-S-H) and/or C-S-H gels. The formation of zeolite phases of sodium alumino-silicate and silica-rich can be attributed to the high solubility of silica and sodium in the alkaline solution under heat curing conditions [22]. We observe little difference between the diffractograms of the raw RUFA and RUFA-based geopolymers. This leads us to conclude that the geopolymers have a geopolymerisation form an amorphous to granular semi-crystalline structure, which is consistence with the previous study [23].
3.3. SEM and EDS analysis

Hitachi S-4100 field emission electron microscope, which has a maximum magnification rate of 100,000, is used for SEM. This instrument uses an electron gun that shoots a beam of electrons. When the electron beam hits the surface of cement-based materials, it produces signals that are transmitted to a cathode ray tube via a signal amplifier and then displayed on a monitor screen, which allows the observation of microscopic crystal phases on the surface of the RUFA geopolymer. By adjusting the magnification rate and moving around the specimen, pictures of the desired microscopic crystal phases can be taken.

Fig. 6 illustrates the microscopic properties of the proposed RUFA-based geopolymer, as follows: (a) SEM image of raw RUFA; (b) SEM photo of paste at 7th days; (c) SEM photo of paste at 28th days. The paste consisted primarily of aluminosilicate gel or geopolymer paste with unreacted RUFA particles. The quantity of geopolymerisation at 28th days is far higher than at 7th days, which may account for the continual increase in compressive strength over that period. The reaction of RUFA with NaOH solution leads gradually to the formation of C-S-H, C-A-S-H, and N-A-S-H gels, which modifies the microstructure of the paste. RUFA geopolymer pastes are presented with multiple unreacted particles. It clearly shows that the inhomogeneous glass-like pastes of the amorphous aluminosilicate gel and the pastes are full of loosely structured RUFA grains of different sizes. This can be expected that if the NaOH content increases, the degree of reaction taking place in a geopolymer forming paste increases. It indicates that it needs more alkali activator with a higher Molar concentration of NaOH to enhance the polymerization process.
The elements analyzed by EDS mainly include Si, Al, Na, Ca and O as shown in Tables 2 and 3. The EDS results from 7th to 28th days reveal the slight gradual decrease in the Na content (from 4.46 % to 4.35 %) as well as gradual increases in the Si/Al ratio (from 2.82 to 3.23) and Ca/Si ratio (from 0.27 to 0.31). The gradual increase in compressive strength can be attributed to the formation of Si-O-Si bonds, which are stronger than Si-O-Al bonds [24-25] and consistent with the XRD results. The previous study [26] reported that the dissolution speed of silicate bonds was faster than that of aluminum bonds; however, the geopolymerization speed of aluminum bonds was faster than that of silicate bonds. It also confirmed that the decrease of the Si/Al ratio may be an indication of the process of geopolymerization [24, 26]. It also showed that gel mostly consists of the phases containing Na-Si-Al bonds and the formation of an alumino-silicate-activated gel by polymerization throughout the surface of the RUFA particles as shown in Figs. 6(b) and 6(c).

| Element | Weight % | Atomic % |
|---------|----------|----------|
| O       | 57.83    | 71.52    |
| Fe      | 2.68     | 0.95     |
| Na      | 4.46     | 3.84     |
| Al      | 6.83     | 5.01     |
| Si      | 19.25    | 13.56    |
| Ca      | 5.26     | 2.60     |
| Mg      | 2.14     | 1.75     |
| K       | 1.54     | 0.78     |
| Total   | 100      | 100      |

Table 3 EDS analysis of RUFA pastes at the age of 28th days

| Element | Weight % | Atomic % |
|---------|----------|----------|
| O       | 56.69    | 70.67    |
| Fe      | 2.88     | 1.03     |
| Na      | 4.35     | 3.77     |
| Al      | 7.19     | 5.32     |
| Si      | 19.13    | 13.59    |
| Ca      | 5.93     | 2.95     |
| Mg      | 2.32     | 1.90     |
| K       | 1.52     | 0.77     |
| Total   | 100      | 100      |

3.4. TG-DTA results

In this investigation, Thermogravimetry (TG) analysis is studied in terms of the weight-loss percentage as an index to define the compositions or formations. The Derivative Thermogravimetry (DTG) curves in Fig. 7 reveals an obvious decrease in mass (3.3 wt.%) as the temperature of the specimens is increased to 200°C. The mass decreases by a further 2.3 wt.% as the
temperature is increased from 200°C to 650°C, which indicates the analyzed bonded water of the Ca(OH)$_2$ phase. A small drop between 200°C to 400°C has found in the DTG curve, which is corresponded to dehydration of C-A-S-H and C-S-H pastes [27-28]. The combination of NaOH and Ca(OH)$_2$ activates the compressive strength development of the RUFA geopolymer. The TG curve remains steady after 650°C due to the swelling of the high silicate secondary phases [29]. The DTA curves reveal the transformation of amorphous aluminum phosphate phase into crystal phases (e.g. tridymite-like molecular structure) at approximately 870 °C. Note that the same phenomenon is observed in geopolymer based on general/traditional fly ash [30]. The DTA curve also reveals that the reaction is endothermic between 300°C and 1200°C, due to the sintering of high silicate secondary phases and extreme densification of the geopolymer pastes [29]. In addition, the loss in weight between 100°C and 700°C from the DTG curve could be due to structurally bonded water in the N-A-S-H gel, which is consistent with the results of the SEM and EDS analysis. It is also an indirect measure of the geopolymerization reaction [31]. It is also revealed that N-A-S-H and C-A-S-H gels prevail in high silicon and aluminum systems in RUFA geopolymer. The compressive strengths at the early stage (7$^{th}$ to 28$^{th}$ days) indicate a very promising performance to be an innovative application in green cementless composites.

Fig. 7 DTG, DTA, and TG curves at 28$^{th}$ days

4. Conclusions

This study reported on a RUFA-based geopolymer created by using 4-M NaOH solution as an activator. The high compressive strength of the proposed geopolymer (13.33 MPa at 28$^{th}$ days) was attributed to the formation of hydration products (calcium- and sodium-based aluminosilicate hydration products). EDS results revealed gradual increases in the Si/Al ratio and Ca/Si ratio over time. Nonetheless, the microstructure presented aluminosilicate gel and unreacted RUFA particles at 28$^{th}$ days. The concentration of the NaOH solution was shown to have a profound effect on strength development and the densification of the geopolymer in future research.

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Conflicts of Interest

The authors declare no conflict of interest.
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