Abstract: Corrosion testing is a very important step in quality control for metal industrial processes. EIS is a versatile procedure for the accelerated evaluation of the anti-corrosion performance of coatings: unlike other standard procedures is generally a non-destructive method. EIS works applying an electrical sinusoidal perturbation with a fixed frequency and measuring electrical impedance $Z$ of the sample. Measuring impedance at different frequencies and analysing the data, it is possible to postulate the structure of an equivalent circuit and extract corrosion resistance data. This approach is commonly used for high-impedance coatings, in this study we explore EIS as well as the OCP measurement, the corrosion current and other techniques to find the best option for low-impedance metallic coatings analysis. The objective of this study is to develop a method to determine corrosion resistance for electroplated goods that can give results as reliable as other more diffuse and traditional destructive corrosion testing techniques (such as corrosion tests in artificial atmosphere ISO 17228, ISO 9227 and ISO 4524) with a non-destructive process and in a fair less amount of time.

Keywords: corrosion; metal coatings; impedance spectroscopy; accelerated testing; salt spray test; EIS; electroplated goods

1. Introduction

The metals are normally exposed to corrosion in atmospheric and aqueous environments; in many industrial sectors metallic corrosion is a well-known problem, for this reason often protective coating are applied to the bare metal. There are many typology of coatings: insulating like plastics, paints and passivation or conductive like galvanic protection or noble element deposition. The uses of these technique ranges from fashion accessories, used in jewellery, bags, belts, etcetera to the automotive sector and in the construction sector where an example can be the bars of the concrete [1–3]. So the development of new alloys or coatings takes on a fundamental aspect for the prevention of corrosion, because all this has negative implications both from an aesthetic point of view and from a point of view of environmental safeguarding, like in the collapse of structures. In order to avoid these problems, it is important to check the corrosion status of the materials.

Nowadays, while for the insulating coating exist electrochemical studies and regulations, for the assessment of corrosion on metal films is carried out through long and destructive tests such as: the 24 h synthetic sweat test [4], the corrosion test with thioacetamide [5] with a duration of 48 h, the
24 h sulphur dioxide test [6], the 24 h damp heat test [7] and the 48 h salt spray test [8]. After performing the tests, the corrosion assessment of the metallic coatings is carried out by a visual comparison. It seems clear that until today, in literature there is no tool that expresses a certain numerical value to identify if a metallic coating has undergone to corrosion.

In facts visual comparison lead only to a quantitative estimation but not a qualitative one. The aim of this work was to test electrochemical methods for the evaluation of corrosion in metal coatings using cumulative charge, Open Circuit Potential (OCP) and Electrochemical Impedance Spectroscopy (EIS) measurements, combined to the application of a stress potential, there may be applications both in industry and in scientific research in the case of conductive film studies [9–16]. Although this technique is already used for the study of materials coated with paints and organic materials [17–19], no study has been carried out on metallic coatings. For this reason, we use methodology that combined the studies on insulating coatings and uncoated metals [20–26], to obtain a faster, alternative, non-visually destructive, quantitative and modulable method for the evaluation of corrosion of metallic coatings, compared to the current methods used in the galvanic industries.

2. Materials and Methods

The samples were prepared on a metal substrate of 3.75 mm × 5 mm brass (copper-zinc 67:33 alloy) plates, 0.25 mm thick. We used commercial electrodeposition solution of copper, white brass, palladium and gold. The sequence of the layers and their thickness is shown in Table 1. We used typical galvanic bath used in jewellery industry with the same ranges of thicknesses and layer sequences [1]. To have a simple system we used only a single non-precious metal layer and one precious layer, since gold is not stable if electroplated directly on copper, due to the solubility of the two metal one in the other and the consequent diffusion over time, gold was electroplated on bronze and palladium on copper. The characteristics of the single layers are the following: copper tends to suffer corrosion, bronze tends to passivation as well as palladium (even if palladium is nobler), and finally gold should be not attacked by corrosion.

Table 1. Composition and thicknesses of the samples.

| Sample | Base Metal | Cu (µm) | Bronze (µm) | Pd (µm) | Au (µm) |
|--------|------------|---------|-------------|---------|---------|
| Au0.3  | Brass      | >5      | 4.46 ± 0.62 | –       | 0.36 ± 0.02 |
| Au0.6  | Brass      | >5      | 4.46 ± 0.62 | –       | 0.74 ± 0.09 |
| Au1.0  | Brass      | >5      | 4.46 ± 0.62 | –       | 1.04 ± 0.07 |
| Pd0.3  | Brass      | >5      | –          | 0.36 ± 0.07 | –       |
| Pd0.6  | Brass      | >5      | –          | 0.67 ± 0.03 | –       |
| Pd1.0  | Brass      | >5      | –          | 1.11 ± 0.06 | –       |

Atomic force microscopy (AFM) (PicoSPM, Molecular Imaging, Tempe, AZ, USA) was used evaluate the roughness of the samples. The measurements were performed in contact mode with a non-conductive Si₃N₄ triangular cantilever (NP-S10, Veeco, Plainview, NY, USA), the cantilever depth was in the 0.4–0.7 µm range and with a force consent of 0.12 N/m, 0.5 V force set point and a speed of 1.21/s.

XRF measurements were performed with a Bowman B Series XRF spectrometer (Schaumburg, IL, USA) using an acquisition time of 60 s, 50 kV tube voltage, 0.8 mA tube current, ad a focal distance of 12 mm and with a collimator of 0.305 mm in diameter.

The electrochemical measurements were performed with a Metrohm Autolab (Utrecht, The Netherlands) PGSTAT204 potentiostat controlled by the NOVA 2.1.3 software. The electrochemical cell consists in a polymethylmethacrylate cylinder, 5 cm in diameter and 1 cm long, with a hole of 11.3 mm in diameter along its axis to contain 1 mL of solution. Measurement was performed with the three electrode setup. The working electrode (our samples) and the counter electrode (Pt foil) were disposed on the two sides of the cylinder and screw sealed with PMMA plug. Through a small lateral hole, the Ag/AgCl/sat. KCl reference electrode is immersed in the solution. The solution contains 3.5 wt % NaCl (Sigma Aldrich, Saint Louis, MO, USA; ≥99%) ant
is buffered at pH 7 with 0.1 M NaH₂PO₄/Na₂HPO₄.

The classical corrosion tests performed was 24 h damp heat ISO 17228:2015 [7], 48 h salt spray ISO 9227:2017 [8] and 24 h sulphur dioxide ISO 4524:2000 [6].

3. Results

The thickness of the coatings was measured using the XRF, the results are shown in Table 1. The samples are named with the primary element in the composition of the final coating, followed by the nominal thickness. The real thickness could be slightly different from the nominal one but the distribution was calculated, in fact six sample of each type was prepared to perform all the planned experiments.

In addition to all the other samples two more were prepared to study the roughness of the substrate. The top layer of the first one was bronze, the substrate of the gold samples, while the other was copper, used as substrate for the palladium samples. We performed the AFM analysis of the substrates (Figure 1) to calculate the RMS of the heights. The founded roughness, 10.0 nm for Cu and 10.3 nm for bronze, was much lower than the lowest deposit in the gold and the palladium samples, confirming that we can consider the deposit as homogeneous.

3.1. Polarization Plots

The first electrochemical characterization of the samples was made performing a long time OCP measurement and a subsequent polarization curve. The OCP was recorded for 16 h on the two substrate (copper and brass) and on the Au1.0 and Pd1.0 samples, to observe the time needed to obtain a stable reading of the corrosion potential $E_{corr}$ and also to observe if there was any drifting. The samples with the higher thickness was chosen for this measurement since we want to isolate the contribution of the top layer alone. The results show (Figure 2) little fluctuations only in the early stages of the measurement and a very light drift for almost all the samples except for the Bronze which keeps increasing its OCP value even after 16 h, in fact bronze start at −0.35 V and stops over −0.10 V, crossing the Cu value at −0.2 V after 2 hours and a half. This is due to the formation of an oxide layer on the surface of the bronze. There is also a cross between Au1.0 and Pd1.0 samples due to the rising of Au1.0, but the variation over the total time is almost negligible.
The polarization curve was performed from $-0.25$ to $+1.3$ V respect to the OCP value (the average on the last 50 points was considered) with a scan speed of 0.4 mV/s. The results are showed in Figure 3, copper reach a high current passive-like zone after the corrosion peak above $0.25$ V; Also bronze own a passive zone after the same potential but with much lower currents; Palladium has a corrosion peak at $0.6$ and over $1$ V the current rises probably due to the oxidation of water; gold has a small corrosion peak near the $E_{corr}$ and another one muck more intense at $0.9$ V, then the water oxidation takes place. Since the bronze suffer a drifting during the OCP measurement, for comparison it was measured another polarization curve of it after an OCP of 1 h. In this case the $E_{corr}$ occurs at much more negative potentials, as expected, but there is also a corrosion peak, at $0.0$ V that was not present the previous measurement.

3.2. Classical Corrosion Tests

All the samples were subjected to three commonly used accelerating corrosion test: damp heat, salt spray and sulphur dioxide (Figure 4). The coatings were smooth and without imperfection that the damp test and sulphur dioxide were not able to attack even the lower thickness. Only the salt spray exhibited some variation before and after the test: gold samples with lower thickness appears more reddish and all the sample present a white line, due to the inclusion of the salt, along the diagonal because of their position in the test chamber, little amount of greenish salt was spotted near the holes (weak points) used to hang up the plates during the sample preparation.
Figure 4. Pictures off all the samples taken before and after the chemical corrosion tests. The three top images in each cell corresponds to the before, while the after is on the bottom, the performed test from left to right are: dump heat, salt spray and sulphur dioxide.

3.3. Electrochemical Tests

The electrochemical test was performed as follow: (1) 5 min OCP measurement; (2) initial EIS measurement from 100 kHz to 10 Hz with 10 samplings per decades; (3) repeated 12 times: application of a stress potential for 30 min and charge sampling; 5 min relaxation and OCP measurement; (4) final EIS measurement from 100 kHz to 10 Hz with 10 samplings per decades

We chose to limit the OCP measurement to 5 minutes to shorten the time needed for the complete test, also considering that the precious top coating reaches a stable goal quickly as seen in Figure 2. We chose two different stress potential: +20 mV and +100 mV respect to the last measured OCP. For all samples after the stresses no visual modification was observed even if their electrochemical behaviour changed. The corrosion was evaluated recording three signals: cumulative charge during the stresses (Figure 5), OCP before and after every stress (Figure 5) and EIS before and after the stresses (Figure 6). For comparison the OCP and the EIS measurement was performed also on the samples subjected to the salt spray, before and after the test. The EIS measurement was reported with the Nyquist plots. The charge gives a qualitative information of the corrosion resistance, in fact high amount of charge may be due to the oxidation of coating and low quantity to resistance but also passivation will lead to make pass only little currents. The amount of charge that pass in the gold samples in higher than for palladium, while the OCP is lower; except for Au1.0, in all the samples the charge is higher for the treatments at +100 mV respect to +20 mV, as expected.

During the test at +20 mV for the Au0.3 the charge tends to increase of one order of magnitude suggesting that modifications are taking place in on the surface, while the other two gold samples exhibit a stable trend. The OCP in these samples remains quite constant during the cycles. Also in the Nyquist plot there is no big changes in the impedance values (Figure 6a), this was ascribed to the resistance of the coating to the corrosion. In all the gold samples, during the +100 mV stress, it was recorded a high amount of charge in the first two cycles, then the current drops, this behaviour was more evident for Au1.0 than the other samples. In this case the OCP goes up in the first two cycles and then falls back in the following ones, meaning that modification takes place on the surface. Also the Nyquist plot shows big changes (Figure 6b) and it is interesting to notice that after the faradic treatment the slope of the curve differs less increasing the thickness. The EIS measurement performed before and after the salt spray test (Figure 6c) are quite similar to those obtained with the +20 mV stress. The palladium samples exhibit a similar trend during both the two stresses and for all the thicknesses: the amount of charge decrease of ten times and the OCP increase during the cycles, this
behaviour was ascribed to the passivation of the metal. This trend is more evident for the thinner samples. The Nyquist plot does not change a lot after the +20 mV treatment on the palladium samples (Figure 6d), while it is evident the effects after the +100 mV ones (Figure 6e): in fact, also in this case the slope of the curve differs less increasing the thickness, in addition to that the final slope of Pd0.3 and Pd0.6 is lower than the initial, indicating corrosion, while in the case of Pd1.0 it is higher, indicating passivation. Similar behaviour emerged in the measurement performed on the samples subjected to the salt spray test (Figure 6f).

**Figure 5.** Cumulative charge plot (a) and OCP variation (b) during the electrochemical stress. Au +20 mV (black), Au +100 mV (red), Pd +20 mV (blue), Pd +100 mV (green); 0.3 samples (solid line), 0.6 samples (dash line), 1.0 samples (dot line).

**Figure 6.** Cont.
Figure 6. Nyquist plot before (solid line) and after (dash line) of the gold samples with +20 mV stress (a), +100 mV stress (b) and salt spray test (c) and the palladium samples with +20 mV stress (d), +100 mV stress (e) and salt spray test (f); 0.3 samples (black), 0.6 samples (red), 1.0 samples (blue).

Another formalism to report impedance variation in corrosion is $\Delta Z\%$. The data were treated considering the relative variation of the logarithmic value of the impedance module (bode plot) at the lowest frequency (10 Hz), named $\Delta Z\%$, as shown by Equation (1):

$$\Delta Z\% = \frac{\log|Z_{\text{after}}| - \log|Z_{\text{before}}|}{\log|Z_{\text{before}}|}$$  \hspace{0.5cm} (1)

Figure 7a,b showed the results of the gold and palladium samples. It is possible to notice that the electrochemical tests are more consistent than the results obtained with the salt spray test, which appear more scattered. The results also confirm the previous deductions: the treatment at +20 does not attack significantly neither of the two metals, in fact $\Delta Z\%$ is close to 0 or even negative. Instead, after the application of +100 mV, all the sample owns an increment in $\Delta Z\%$. However, there are not significant differences for different thicknesses of the same metal, this could be due to the fact that using $\Delta Z\%$ the information about the real (resistance) and the imaginary (reactance) parts of the impedance are lost so cannot be distinguished between corrosion and passivation.

Figure 7. Comparison after the +20 mV stress (black), +100 mV stress (red) and salt spray test (blue) of the calculated $\Delta Z\%$ value for the gold (a) and the palladium (b) samples.

4. Conclusions

In this paper we evaluate the corrosion of metal coatings using electrochemical methods. EIS is a common technique used to test insulating coatings and paints but not many studies are present to test the resistance of metal films. Nowadays this kind of samples are tested with long and destructive methodology. In this study we experiment three different thicknesses of commonly electroplated alloys in precious galvanic industry: gold films on bronze and three of palladium on copper, namely a noble metal deposited on a not precious one. The results were also compared with more classical...
tests. The proposed method is not visually destructive, lead to numerical results (instead of visual comparison), the stress could be easily modulated and needs about 8 h (compared to the 24 or 48 h of chemical tests) but could be even faster if the number of cycles is reduced or the potential stress increased. On the other hand, the interpretation of the results is not immediate, we evaluate the cumulative charge and OCP trends, Nyquist plot variation and ΔZ%. For this reason, even if this technique shows high potentials, more study must be done on samples with different thickness and composition to find which parameter, or parameters, are more significant for a simple interpretation of metal film corrosion.

Author Contributions: Conceptualization, W.G.; Methodology, W.G.; Investigation, F.B.; Data Curation, W.G. and F.B.; Writing—Original Draft Preparation, W.G. and F.B.; Writing—Review & Editing, P.M., F.G., M.P., E.S. and A.D.L.; Supervision, W.G. and D.B.; Project Administration, M.I.; Funding Acquisition, M.I.

Funding: This research was funded by Regione Toscana POR CreO FESR 2014-2020—azione 1.1.5 sub-azione a1 Bando 2 “Progetti di ricerca e sviluppo delle MPMI” which made possible the project “Gioielli in Argento Da Galvanica Ecologica e Tecnologica” (GADGET) and “Tecnologia al plasma per l’industria del lusso: una manifattura innovativa nel comparto accessori in ottica 4.0” (THIN FASHION).

Acknowledgments: The authors acknowledge Metrohm s.r.l. for the technical support and the devices made available.

Conflicts of Interest: The authors declare no conflict of interest.

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