Fabrication and Physical Properties of Single-Crystalline Beta-FeSi$_2$ Nanowires

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Abstract

In this study, self-catalyzed $\beta$-FeSi$_2$ nanowires, having been wanted but seldom achieved in a furnace, were synthesized via chemical vapor deposition method where the fabrication of $\beta$-FeSi$_2$ nanowires occurred on Si (100) substrates through the decomposition of the single-source precursor of anhydrous FeCl$_3$ powders at 750–950 °C. We carefully varied temperatures, duration time, and the flow rates of carrier gases to control and investigate the growth of the nanowires. The morphology of the $\beta$-FeSi$_2$ nanowires was observed with scanning electron microscopy (SEM), while the structure of them was analyzed with X-ray diffraction (XRD) and transmission electron microscopy (TEM). The growth mechanism has been proposed and the physical properties of the iron disilicide nanowires were measured as well. In terms of the magnetization of $\beta$-FeSi$_2$, nanowires were found to be different from bulk and thin film; additionally, longer $\beta$-FeSi$_2$ nanowires possessed better magnetic properties, showing the room-temperature ferromagnetic behavior. Field emission measurements demonstrate that $\beta$-FeSi$_2$ nanowires can be applied in field emitters.

Keywords: $\beta$-FeSi$_2$, Nanowire, CVD, Ferromagnetic, Field emission

Introduction

As the dimension of CMOS devices is down to the nanoscale, metal silicide technology will be even more significant; the substrate of many photonics and microelectronics devices has been silicon. Transition-metal silicides have been studied extensively owing to their outstanding properties, including low resistivity, and great stability [1–5]. For instance, CrSi$_2$, $\beta$-FeSi$_2$, and MnSi are suitable as thermoelectric materials due to their narrow energy gap and great thermostability [6]; NiSi, CoSi$_2$, and TiSi$_2$ are frequently utilized as materials of the metal gate for decreasing the resistance [7].

With excellent properties, such as high compatibility and low defect density, one-dimensional nanostructures are promising for current and future microelectronic devices [8], drawing widespread attention not only from academic studies but also from industry applications [9]. For the past few years, growth kinetics of various metal silicide nanowires, including transition-metal silicides and rare-earth silicides, has been studied [10–14].

There are different phases for iron disilicides [15–19], among which, the unusual characteristics of $\beta$-FeSi$_2$ is particularly fascinating. As previously reported, $\beta$-FeSi$_2$ nanowires were demonstrated to have important applications in the field of communication [20]; unfortunately, over the many years, few have been able to successfully repeat the fabrication of $\beta$-FeSi$_2$ nanowires with chemical vapor deposition. For $\beta$-FeSi$_2$, the room temperature equilibrium phase, the potential applications in light emitters, and infrared detectors for silicon-based optoelectronics are attributed to its direct bandgap. In this work, we report direct growth and structural characterization of the single crystalline $\beta$-FeSi$_2$ nanowires via a chemical vapor deposition method. The as-synthesized $\beta$-FeSi$_2$ nanowires...
exhibited the room-temperature ferromagnetic behavior. Field emission measurements show that the β-FeSi$_2$ nanowires are great field emission materials.

**Methods**

In this study, we synthesized β-iron disilicide nanowires using chemical vapor deposition with anhydrous FeCl$_3$ powder as a precursor, silicon (100) substrates, and Ar carrier gas but without any catalysts. Silicon substrates were cleaned with 3 %-buffered HF and put in the downstream zone of the furnace; anhydrous FeCl$_3$ powder was placed in an alumina boat upstream from the substrates, the temperature range of which was 750 ~ 950 °C. We carefully varied temperatures, duration time, and the flow rates of carrier gases in order to realize the factors that influenced the growth of β-iron disilicide nanowires. We utilized scanning electron microscopy (SEM) to investigate the morphology of the β-FeSi$_2$ nanowires; X-ray diffraction (XRD) and transmission electron microscopy (TEM) studies were conducted for structural identification. In addition, characteristics such as magnetism and field emission property were measured. The magnetic property measurements of the β-FeSi$_2$ nanowires were conducted by the Superconducting Quantum Interference Device (SQUID) with the VSM option, while field emission property was measured by Kiethly-237.

**Results and Discussion**

We explored the parameters that could affect the growth of the β-iron disilicide nanowires. Firstly, different gas flow rates were investigated from 50 to 200 sccm as shown in the SEM images of Fig. 1a–c. Figure 1a reveals the gas flow rate at 50 sccm, where we found lots of nanowires with the diameters of 40 nm and lengths of 10 μm. Figure 1b shows the gas flow rate at 80 sccm, where some nanowires but the amount was reduced. In Fig. 1c showing the gas flow rate at 120 sccm, there were even fewer nanowires formed. According to the results, the amount of nanowires decreased with the increase of the gas flow rates. When the nanowires grow, the precursor, FeCl$_3$, should be carried to the downstream zone of the tube furnace and react with the Si substrate by a carrier gas. At higher gas flow rates, it may be difficult for the nanowires to grow. Based on the chemical vapor deposition mechanisms, there were generally five steps in the deposition process, namely, (1) diffusion of reactants to the surface, (2) absorption of reactants at the surface, (3) chemical reaction at the surface, (4) desorption of products from the surface, and (5) diffusion of products from the surface. The slowest step determines the rate of the CVD reaction. If (1) or (5) is the slowest step, it is mass transfer-controlled. If (2), (3), or (4) is the slowest step, it can be called “surface reaction-controlled.” At low temperatures and slow gas flow rates, the surface chemical reaction is slower than reactant diffusion; thus, it is surface reaction-controlled. When it is surface reaction-controlled, variation in the film thickness across the wafer will depend on the distribution of temperature, and thin film tends to form. However, our purpose is to grow nanowires; therefore, we should avoid surface reaction-controlled reaction. On the other hand, it is mass transfer-controlled at high temperatures and low gas flow rates. When it is mass transfer-controlled, the rate of the top obtained reactants is faster than that at the sidewall since the axial growth is faster than the radial growth; as a result, nanowires tend to form, and thus, we obtained many nanowires with decreasing gas flow rates. Therefore, mass transfer-controlled reaction is necessary for nanowire growth.

The second parameter we investigated was different growth temperatures as shown in the SEM images of Fig. 1d–f. Figure 1d reveals the growth temperature at 750 °C, where there were some nanowires but their lengths and diameters were short and small. Figure 1e shows the growth temperature at 850 °C, where we found lots of nanowires with the diameters of 40 nm and lengths of 10 μm. When we increased the growth temperature to 950 °C as shown in Fig. 1f, nanowires became nanorods due to more deposition of precursors. The third parameter we investigated was the duration time; Fig. 1g–i shows the SEM images for 1 h, 2 h, and 5 h. Generally, we found longer nanowires with the increasing duration time. After more than 5 h, the morphology of the nanowires would not change significantly, which may be attributed to the fact that the precursor had been completely consumed.

To identify the structure of the nanowires, we conducted X-ray diffraction (XRD) and transmission electron microscopy (TEM) analysis as shown in Fig. 2. All the peaks in the corresponding XRD spectrum could be indexed to the structure of orthorhombic β-FeSi$_2$ phase as shown in Fig. 2a. Figure 2b is a TEM image showing a single-crystalline β-FeSi$_2$ nanowire; Fig. 2c is the high-resolution TEM image with the inset of the corresponding fast Fourier transform (FFT) diffraction pattern, showing that the β-FeSi$_2$ nanowire has an orthorhombic structure with [200] growth direction and that the interplanar spacings of planes (200) and (111) are 0.493 nm and 0.482 nm, respectively.

The growth mechanism in our experiment may involve two reactions to produce β-FeSi$_2$ nanowires as shown in Fig. 3. In the first reaction, evaporative FeCl$_3$ was carried to the furnace downstream, reacting with...
the Si substrate to form $\beta$-FeSi$_2$ nanoparticles and by-products of SiCl$_4$ with $\beta$-FeSi$_2$ nanoparticles appearing increasingly. In the second reaction, SiCl$_4$ from the first reaction would also react with the precursor of evaporative FeCl$_3$ and form $\beta$-FeSi$_2$ and Cl$_2$. With Cl$_2$ carried out by Ar gas, we gradually obtained $\beta$-FeSi$_2$ nanowires from both the first and second reactions. The growth mechanism was VS because we did not observe catalyst-like metal droplets at the front end of the nanowire. The synthesis via VLS mechanism requires a catalyst; however, no catalyst was used in the experiment. To further investigate the growth mechanism, we tried hydrogen, which may have a reduction effect; still, no metal catalytic droplet was formed. Therefore, we demonstrate that the growth mechanism was VS.

The magnetization of $\beta$-FeSi$_2$ was interesting with different dimensions. It has been found to exhibit superparamagnetism in nanoparticles, even though no magnetic ordering occurs in bulk [21], while in the case of $\beta$-FeSi$_2$ thin film, ferromagnetism was found only at temperatures below 100 K [22]. The ferromagnetic behavior of $\beta$-FeSi$_2$ nanowires may be due to the large specific surface area of the nanowire, leading to many unpaired iron atoms on the surface. Additionally, some strain and defects generated during the growth process could be another factor contributing to the ferromagnetism. To examine the magnetic properties of the grown $\beta$-FeSi$_2$ nanowires, the magnetic properties were measured using the Superconducting Quantum Interference Device (SQUID) with VSM option.

Figure 4a is the magnetic response only from the silicon substrate, which clearly shows diamagnetic behavior; we subtracted the magnetism of the silicon substrate for all the following magnetism of $\beta$-FeSi$_2$ nanowires. The magnetization curve of the $\beta$-FeSi$_2$ nanowires was growing in 2 h as shown in Fig. 4b. The nonlinear hysteresis loop curve shows that the $\beta$-FeSi$_2$ nanowires exhibited

![SEM images of $\beta$-FeSi$_2$ nanowires at different parameters. At different gas flow rates: a 50 sccm, b 80 sccm, and c 120 sccm. At different temperatures: d 750 °C, e 850 °C, and f 950 °C. At different duration times: g 1 h, h 2 h, and i 5 h](image-url)
ferromagnetic behavior at room temperature. The coercivity was about 264 Oe. Larger saturation magnetization was found at 2 K because of the decreasing thermal fluctuation. Due to the reduced coordination of the surface iron atoms, or the strain and structural defects in the crystal, \( \beta \)-FeSi\(_2\) nanowires grown here were found to be ferromagnetic [23]. Figure 4c shows the magnetization curve of the longer \( \beta \)-FeSi\(_2\) nanowires growing in 5 h. From shorter to longer nanowires, the coercivity increased from 264 to 345 Oe at 300 K, and even to 575
Oe at 2 K; saturation magnetization was raised more as well. It was confirmed that the longer nanowires possessed better magnetic properties. Temperature-dependent field cooling (FC) and zero-field cooling (ZFC) magnetization measurements are shown in Fig. 4d, where the magnetization curve did not drop to zero, showing that the Curie temperature of $\beta$-FeSi$_2$ NWs was higher than room temperature. The ZFC and FC curves of $\beta$-FeSi$_2$ NWs did not overlap; the temperature of curve separation is called blocking temperature ($T_b$), indicating that a large magnetic anisotropy energy barrier distribution existed [24]. When the temperature was lower than $T_b$, the magnetic anisotropy energy was larger than the thermal fluctuation. As a result, grains were blocked and not impacted by the applied magnetic field; thus, the magnetization was observed.

To explore the field emission properties, we conducted the field emission measurements for the $\beta$-FeSi$_2$ nanowires. The sample was measured in a vacuum chamber at $\sim 10^{-6}$ torr. Figure 5 shows the current density ($J$) - field ($E$) plot with $\beta$-FeSi$_2$ nanowires of different lengths. According to the Fowler–Nordheim (F–N) plot and the Fowler–Nordheim equation:

$$J = \left(\frac{A\beta E^2}{\phi}\right) \exp\left(-\frac{B\phi^{3/2}}{\beta E}\right),$$

where $J$ is the current density, $E$ is the applied electric field, and $\phi$ is the work function; the inset reveals the $\ln(J/E^2) - 1/E$ plot. The field enhancement $\beta$ was calculated to be 1060 from the slope of $\ln(J/E^2) = \ln(A\beta^2/\phi) - B\phi^{3/2}/\beta E$, and $\beta$ increased from 1060 to 2367 with the increasing length of nanowires, demonstrating that longer $\beta$-FeSi$_2$ NWs had better field emission properties than shorter ones, and that $\beta$-FeSi$_2$ NWs could be great field emission materials.
Conclusions

β-FeSi₂ nanowires were successfully synthesized with a CVD method. Processing parameters, including temperature, gas flow rate, and duration time were investigated for their effect on the nanowire growth. The growth mechanism has been proposed. Unlike bulk and thin-film β-FeSi₂, the as-synthesized β-FeSi₂ nanowires exhibited room-temperature ferromagnetic behavior. Field emission measurements demonstrate the β-FeSi₂ nanowires as potential field emitting materials.

Abbreviations
SEM: Scanning electron microscopy; XRD: X-ray diffraction; TEM: Transmission electron microscopy; CMOS: Complementary metal-oxide-semiconductor; CVD: Chemical vapor deposition; FFT: Fast Fourier transform; HRTEM: High-resolution transmission electron microscope; VLS: Vapor–liquid–solid method; VSM: Vibrating sample magnetometer; SQUID: Superconducting quantum interference device; FC: Field cooling; ZFC: Zero-field cooling; T_b: Blocking temperature

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Authors’ Contributions
C.Y. Yang and K.C. Lu conceived the study and designed the research. C.Y. Yang conducted the experiments. C.Y. Yang, S.M. Yang, Y.Y. Chen, and K.C. Lu wrote the manuscript. The authors read and approved the final manuscript.

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Availability of Data and Materials
The data supporting our findings are included in the article.

Competing Interests
The authors declare that they have no competing interests.

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References
1. Lin JY, Hsu HM, Lu KC (2015) Growth of single-crystalline nickel silicide nanowires with excellent physical properties. CrystEngComm 17:1911–1916
2. Hsu HF, Tsai PC, Lu KC (2015) Single-crystalline chromium silicide nanowires and their physical properties. Nanoscale Res Lett 10:50
3. Huang WJ, Chen YY, Hsu HM, Lu KC (2018) Single crystalline iron silicide and beta-iron disilicide nanowires formed through chemical vapor deposition. Materials 11:2384
4. Lu KC, Wu WW, Ouyang H, Lin YC, Huang Y, Wang CW, Wu ZW, Huang CW, Chen LJ, Tu KN (2011) The influence of surface oxide on the growth of metal/semiconductor nanowires. Nano Lett 11:2753–2758
5. Wu WW, Lu KC, Wang CW, Hsieh HY, Chen SY, Chou YC, Yu SY, Chenand LJ, Tu KN (2010) Growth of multiple metal/semiconductor nanoheterostructures through point and line contact reactions. Nano Lett 10:3984–3989
6. Rowe DM (1994) CRC Press. 50:11235
7. Riffel M, Gross E, Stohrer U (1995) Electrical contacts for FeSi₂ and higher manganese silicide thermoelectric elements. J Mater Sci Mater Electron 6:182–185
8. Jin S, Andrew L, Schmitt ZL, Song Y, Szcech J (2006) Synthesis characterization and physical properties of transition metal silicide nanowires. SPIE 6370:1117–1126
9. Preinesberger C, Vandre S, Kalka T, Dahne-Prietsch M (1998) Formation of dysprosium silicide wires on Si(001). J Appl Phys 31:143–145
10. Chen Y, Ohtberg DAA, Medeiros-Ribeiro G, Chang YA, Williams RS (2000) Self-assembled growth of epitaxial erbium disilicide nanowires on silicon (001). Appl Phys Lett 76:4004–4006
11. Chen Y, Ohtberg DAA, Williams RS (2002) Nanowires of four epitaxial hexagonal silicides grown on Si(001). J Appl Phys 91:3213–3218
12. Preinesberger C, Becker SK, Vandé S, Kalka T, Dahne M (2002) Structure of DySi₂ nanowires on Si(001). J Appl Phys 91:1695–1697
13. Stevens M, He Z, Smith DJ, Bennett PA (2003) Structure and orientation of epitaxial titanium silicide nanowires determined by electron microdiffraction. J Appl Phys 93:5670–5674

Fig. 5 The field emission plot of β-FeSi₂ NWs with different dimensions. The inset shows the corresponding In(J/2²)-1/E plot.
14. Nogami J, Liu BZ, Katkov MV, Ohbuchi C, Birge NO (2001) Self-assembled rare-earth silicide nanowires on Si(001). Phys Rev B 63:233305
15. Borisenko VE (2000) Thin Film Silicide Formation. Semiconducting Silicides. Springer Verlag. p. 81-136.
16. Kanel HV, Onida N, Simninghaus H, Mullergubler E, Goncalvesconto S, Schwarz C (1993) Epitaxial Phase Transitions in the Iron Silicon System. Appl Surf Sci 70(71):559–561
17. Desimoni J, Bernas H, Behar M, Lin XW, Washburn S, Lilientalweber Z (1993) Ion-beam synthesis of cubic FeSi2. Appl Phys Lett 62:306–308
18. Alvarez J, Hinairejos JJ, Michel EG, Miranda R (1993) Determination of the Fe/Si(111) phase diagram by means of photoelectron spectroscopies. Surf Sci 287:490–494
19. Christensen NE (1990) Electronic-structure of β-FeSi2. Phys Rev B 42:7148–7153
20. Leong D, Harry M, Reeson KJ, Homewood JP (1997) A silicon/iron-diliscilicide light-emitting diode operating at a wavelength of 1.5 μm. Nature 387:686–688
21. Chen YY, Lee PC, Tsai CB, Neelleshwar S, Wang CR, Ho JC, Hamdeh HH (2007) Chemical disorder-induced magnetism in FeSi2 nanoparticles. J Appl Phys 91:251907
22. Valassiades O, Dimitriadis CA, Werne JH (1991) Galvanomagnetic behavior of semiconducting FeSi2 films. J Appl Phys 70:890–893
23. Seo K, Varadwaj KSK, Mohanty P, Lee S (2007) Magnetic properties of single-crystalline CoSi nanowires. Nano Lett 7:1240–1245
24. Bagkar N, Seo K, Yoon H, In J, Jo Y, Kim B (2010) Vertically aligned single-crystalline ferromagnetic Ni3Co nanowires. Chem Mater 22:1831–1835

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