Electronic structure of transferred graphene/h-BN van der Waals heterostructures with nonzero stacking angles by nano-ARPES

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Abstract

In van der Waals heterostructures, the periodic potential from the Moiré superlattice can be used as a control knob to modulate the electronic structure of the constituent materials. Here we present a nanoscale angle-resolved photoemission spectroscopy (Nano-ARPES) study of transferred graphene/h-BN heterostructures with two different stacking angles of 2.4° and 4.3° respectively. Our measurements reveal six replicas of graphene Dirac cones at the superlattice Brillouin zone (SBZ) centers. The size of the SBZ and its relative rotation angle to the graphene BZ are in good agreement with Moiré superlattice period extracted from atomic force microscopy (AFM) measurements. Comparison to epitaxial graphene/h-BN with 0° stacking angles suggests that the interaction between graphene and h-BN decreases with increasing stacking angle.

keywords: graphene/h-BN, van der Waals heterostructure, Moiré potential, nanoscale angle-resolved photoemission spectroscopy (nano-ARPES)

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I. INTRODUCTION

FIG. 1: Moiré pattern in graphene/h-BN heterostructure. (a) Schematic drawing of the Moiré pattern in graphene/h-BN heterostructure with 0° stacking angle. (b) Schematic drawing of the Moiré pattern in graphene/h-BN heterostructure with 4.3° stacking angle. The inset schematically shows the relative stacking angle between graphene and h-BN. (c) Calculated Moiré pattern period $\lambda$ and rotation angle $\theta$ as a function of the stacking angle $\phi$. (d) Schematic drawing of the Brillouin zones of graphene and Moiré pattern. The $\Gamma$ (black circles), $\bar{\Gamma}$ (K) (pink circles) points of graphene Brillouin zone and the six nearest SBZ centers (green circles) are shown.

Van der Waals heterostructures are designed heterostructures made by stacking different two dimensional materials which are held together through weak van der Waals interaction [1]. Such heterostructures not only broaden the range of materials that can be assembled and investigated, but also provide an important playground for discovering new properties different from the constituent materials and for realizing new quantum phenomena. In the past few years, graphene/h-BN has emerged as a model van der Waals heterostructure. It is an ideal system for making high quality graphene devices, with reduced ripples and higher mobility [2,3]. Moreover, the moiré superlattice induced by the lattice mismatch and crystal orientation can significantly modify the electronic properties of graphene, leading to various novel quantum phenomena including the self-similar Hofstadter butterfly states [4–7] and topological currents [8]. There are also major changes in the electronic properties, e.g. emergence of second-generation Dirac cones (SDCs) [4–7], renormalization of the Fermi velocity [9–12], gap opening at the Dirac point [6,8,13–15] and gate-dependent pseudospin mixing [16]. Although graphene/h-BN heterostructure has been studied by various transport and STM measurements [4,7,17], direct angle-resolved photoemission spectroscopy (ARPES)
FIG. 2: Preparation of transferred graphene/h-BN heterostructure with 2.4° stacking angle. (a) Exfoliated graphene flake on a suspended PMMA/PVA film. (b) Exfoliated h-BN flake on SiO$_2$/Si substrate. (c) Transferred graphene/h-BN heterostructure. (d) Optical image of the whole sample after deposition of Pt electrode. The location of target graphene/h-BN flake is highlighted by red circle. The square size is 500µm×500µm. (e) Zoom-in of graphene/h-BN sample being measured. The white scale bars shown in above panels are 20 µm. (f) Phase profile along the white line in the inset AFM image. The inset shows the AFM image of the moiré pattern after high-pass-filtered inverse fast Fourier transformation.

measurements of the modulated electronic band structure have been missing.

Hexagonal Boron nitride (h-BN) shares similar honeycomb lattice structure with graphene, yet with a δ ≈1.8% larger lattice constant. The breaking of the inversion symmetry by distinct boron and nitrogen sublattices leads to a large band gap (5.97 eV) in the π band, which is in sharp contrast to the gapless Dirac cones in graphene. By placing graphene atop h-BN, the different lattice constant and relative stacking angle φ between
FIG. 3: Preparation of transferred graphene/h-BN heterostructure with 4.3° stacking angle. (a) Exfoliated graphene flake on a suspended PMMA/PVA film. (b) Exfoliated h-BN flake on SiO$_2$/Si substrate. (c) Transferred graphene/h-BN heterostructure. (d) Optical image of the whole sample after deposition of Pt electrode. The location of target graphene/h-BN flake is highlighted by red circle. The square size is 500 µm × 500 µm. (e) Zoom in of graphene/h-BN sample being measured. The white scalebars shown in above panels are 20 µm. (f) Phase profile along the white line in the inset AFM image. The inset shows the AFM image of the moiré pattern after high-pass-filtered inverse fast Fourier transformation.

graphene and h-BN lead to moiré pattern. The moiré pattern periodicity is

$$\lambda = \frac{(1 + \delta)a}{\sqrt{2(1 + \delta)(1 - \cos \phi) + \delta^2}} \quad (1)$$

where a is the lattice constant of graphene. The relative rotation angle $\theta$ of the moiré pattern with respect to the graphene lattice is given by

$$\tan \theta = \frac{\sin \phi}{(1 + \delta) - \cos \phi} \quad (2)$$

Figure 1(a,b) shows the dependence of the moiré periodicity and rotation angle on $\phi$. The moiré periodicity quickly decreases with increasing $\phi$ and the rotation angle of the moiré
pattern also strongly depends on $\phi$. For example, when $\phi$ changes from 0° to 4.3°, the moiré periodicity $\theta$ changes from 14 nm to 3.2 nm, and the relative angle changes from 0° to 74.5°. The superlattice Brillouin zones (SBZs) are also rotated with respect to the Brillouin zone of graphene due to the rotated moiré pattern which is shown in Fig.1(d) schematically. One expected result of the induced moiré superlattice potential is the formation of first-generation Dirac cones (FDCs) which occur at the same energy level as the original Dirac cone (ODC) yet translated by the reciprocal lattice vector of the moiré pattern $G = \frac{2}{\sqrt{3}} \frac{2\pi}{\lambda}$. Furthermore, due to the induced moiré potential, second-generation Dirac cones (SDCs) can emerge at energies different from the ODC [7, 10, 11, 18], and they are critical for the realization of self-similar Hofstadter butterfly states under applied magnetic field.

Graphene/h-BN heterostructures can be prepared by directly growing epitaxial graphene on h-BN substrate using plasma-enhanced chemical vapor deposition (PE-CVD) [19] or by transferring exfoliated graphene onto the h-BN substrates [2, 3]. While the stacking angle in PE-CVD grown samples is fixed to 0° [19], the stacking angle in transferred graphene/h-BN samples can be carefully aligned and is widely tunable [15, 17, 20]. With increasing stacking angle, the interaction between graphene and h-BN is expected to become weaker and commensurate-incommensurate transition has been reported [15]. The PE-CVD heterostructures with 0° stacking angle and sample size up to a few hundred micrometers ($\mu$m) have been recently studied by regular ARPES, and SDCs have been reported [21]. However, until now direct experimental results on the modulated band structure in graphene/h-BN with nonzero stacking angles have been missing. The challenge is related to the small sample size of a few micrometers ($\mu$m), which is much smaller than the typical ARPES beam size of 50-100 $\mu$m. By using nano-ARPES with beam size of $\sim$100 nm, we are able to obtain the electronic structure of transferred graphene/h-BN heterostructures with nonzero stacking angles for the first time. Here we report the electronic structure of such heterostructures with stacking angles of $\sim$ 2.4° and $\sim$ 4.3° respectively, and reveal the FDCs induced by the moiré superlattice potential.

II. METHODS

Transferred graphene/h-BN heterostructures were prepared using similar methods as reported [2, 3]. The entire process is shown in Figure 2. First, single layer graphene was exfoli-
FIG. 4: Identification of graphene/h-BN heterostructure with 2.4° stacking angle from angle-integrated intensity maps. (a) Optical image of sample #1. The square size is 200 µm × 200 µm. (b) Two characteristic angle-integrated intensity curves from the Pt(black) and SiO₂/Si(blue) as indicated in (a). (c) Spatial map obtained by integrating the intensity of the blue shadow area shown in (b). (d) Zoom-in of sample area from (c) with finer steps. (e) Intensity map around the graphene/h-BN flake by integrating the grey shadow area near the Fermi energy shown in (b).

ated on polymethyl methacrylate(PMMA, MicroChem, A6, 950K)/polyvinyl alcohol(PVA, Sigma-Aldrich) stack (Fig. 2(a)). Single crystals of h-BN were grown using the method described before [22] and h-BN flakes were exfoliated on a SiO₂/Si wafer (Fig. 2(b)). Then the exfoliated graphene was transferred atop exfoliated h-BN flakes (Fig. 2(c)). Pt electrode was deposited on one side of SiO₂/Si substrate (Fig. 2(d,e)) to avoid charging effect during ARPES measurements. Atomic Force Microscopy (AFM) measurements were performed to
FIG. 5: **First-generation Dirac cones in the intensity maps of sample #1 with 2.4° stacking angle.** (a-h) Intensity maps at $E_F$, -100, -200, -300, -400, -500, -600 and -700 meV respectively. The Brillouin zones of graphene and moiré pattern are indicated by blue solid and black dashed lines. Confirm the existence of moiré pattern. The extracted moiré periodicity can also be used to determine the stacking angle between graphene and h-BN. Figure 2(f) shows the extracted phase profile for sample #1. The moiré periodicity is estimated to be 5.3 ± 0.2 nm and the stacking angle is 2.4° ± 0.2° calculated from equation (1). The optical images for sample #2 during the sample preparation are shown in Fig.3(a-e). AFM measurements were also performed to verify the existence of moiré pattern afterwards. Figure 3(f) shows the extracted phase profile from AFM image. The moiré pattern periodicity and the stacking angle are extracted to be 3.2 ± 0.3 nm and 4.3° ± 0.3° respectively.

Nano-ARPES measurements were performed at the ANTARES beamline of the SOLEIL synchrotron. ANTARES beamline is equipped with two Fresnel zone plates (FZPs) to focus the beam size down to ≈ 120 nm. There are two sets of motor systems to change the sample position, mechanical motors SZ and ST which have large motion range with ≈ 5 $\mu$m resolution and piezoelectric motors PIX and PIY which have smaller motion range and better spatial resolution ≈ 5 nm. Samples were mounted on a nanopositioning stage which was placed at the coincident focal point of the electron analyzer and the FZPs. ANTARES can operate in two modes, the imaging mode where the photoelectron spectra (angle integrated or angle resolved) are collected by changing the sample position to create two-dimensional
FIG. 6: Detailed band dispersions of sample #1 with 2.4° stacking angle. (a) Fermi surface map. The Brillouin zone of the moiré superlattice is indicated by grey dashed hexagon. The lines mark the position of cuts shown in b to f. (b-c) Band dispersions along the cuts shown in (a). The dispersions from FDCs are indicated by black arrows. (d-f) Band dispersions along the cuts across the ODC and FDCs as shown in (a). The intersection positions of ODC and FDCs are indicated by black arrows. (g) MDCs at $E_F$ from (d-f). The momentum positions of first-generation Dirac points are indicated by black arrows.

Images of electronic states of interest, and the spectroscopy mode where the detailed band dispersions are measured at fixed sample position. The data were recorded with Scienta R4000 analyzer with photon energy of 100 eV using horizontal linear polarized light. The samples were annealed at $\approx 250 ^\circ C$ until clean dispersions were obtained. The samples are kept at 80 K with vacuum better than $2 \times 10^{-10}$ Torr during the ARPES measurements.
III. RESULTS AND DISCUSSION

We first used the imaging mode to locate the small target graphene/h-BN flake on SiO$_2$/Si substrate for sample #1 (red circle in Fig.2(d)). The angle-integrated intensity curves show two typical spectra as shown in Fig.4(b). The black curve with stronger intensity near $E_F$ is attributed to the Pt electron, and the blue curve with suppressed intensity near $E_F$ and a peak at $\approx -13$ eV is attributed to the bare SiO$_2$/Si substrate [23]. By integrating the spectral weight of the blue shadow area which is characteristic of SiO$_2$/Si (Fig.4(b)), the spatial map (Fig.4(c)) shows similar shape with SiO$_2$/Si in the optical image (Fig.4(a)). This confirms our assignment of the two types of spectra. The zoom-in intensity map around the graphene/h-BN region (Fig.4(d)) allows to further distinguish different parts (Pt, graphene, h-BN and SiO$_2$/Si) clearly. Moreover, by integrating the spectral weight near the Fermi energy (grey shadow area in Fig.4(b)) to have a better contrast between graphene/h-BN and bare h-BN, the graphene flake can be clearly identified (Fig.4(e)).

**FIG. 7:** First-generation Dirac cones shown in the intensity maps of sample #2 with 4.3° stacking angle. (a-h) Intensity maps at $E_F$, -100meV, -200meV, -300meV, -400meV, -500meV, -600meV and -700meV respectively. The Brillouin zones of graphene and moiré pattern are indicated by blue solid and black dashed lines.
After locating target graphene/h-BN flake, we used the spectroscopy mode to probe the modulated band structure of transferred graphene/h-BN. Figure 5 shows the intensity maps at energies from $E_F$ to -700 meV. At $E_F$, six cloned FDCs are observed around the ODC, consistent with our previous ARPES studies on PE-CVD graphene/h-BN heterostructures [21]. The corresponding SBZ (black dashed hexagon) is rotated with respect to the Brillouin zone of graphene (blue solid line) by $\approx 6^\circ$, which is consistent with the calculated rotation angle for the moiré pattern $\theta \approx 66^\circ$ (effectively $6^\circ$ in the map due to the six fold symmetry) from the stacking angle $\phi \approx 2.4^\circ$ by equation (2). The size of the pockets becomes larger with decreasing energy, consistent with the conical dispersions expected. Figure 6 further shows detailed cuts around the ODC and FDCs to reveal the modulated band structures. The dispersions from the FDCs are obvious on both sides of ODC in Fig. 6(b,c). Panels (d-f) show the dispersions along the cuts crossing the ODC and FDCs simultaneously. From the momentum distribution curves (MDCs) at the Fermi energy (Fig. 6(g)), we extract the momentum separation between the ODC and FDCs to be $0.134 \pm 0.02 \, \text{Å}^{-1}$ which is consistent with the moiré pattern periodicity inferred from AFM measurements. Using a Fermi velocity of $7.37 \, \text{eV} \cdot \text{Å} (1.13 \times 10^6 \, \text{m/s})$, the intersection point for $2.4^\circ$ heterostructure is estimated to be at -493 meV. From ARPES measurements, the crossing point is measured to be at -490 meV (pointed by black arrows in Fig. 6(d-f)), in agreement with the estimation, however in contrast with our previous studies on $0^\circ$ aligned graphene/h-BN [21], no obvious signatures of SDCs are observed from the constant energy maps and band dispersions. This is possibly due to the weaker interaction between graphene and h-BN when increasing the stacking angle between graphene and h-BN.

We applied the same method to locate the target graphene/h-BN flake for sample #2. Then spectroscopy mode was used to probe the detailed band structure. Figure 7 shows the intensity maps at from $E_F$ to - 700 meV. The signal from FDCs is weak at $E_F$ and becomes more clear at -100 meV. This could be attributed to even weaker interaction between graphene and h-BN with an increasing stacking angle. The corresponding SBZ (black dashed hexagon) is rotated by $\approx 15^\circ$ with respect to the Brillouin zone of graphene (blue solid lines in Fig. 7(b)), which is consistent with the calculated moiré pattern rotation angle of 74.5° from the stacking angle of 4.3°. Figure 8 shows the detailed modulated band dispersions around the ODC and FDCs. The dispersions from the FDCs are weak but still detectable on both sides of the ODC (Fig. 8(b-d)). Panel (e-g) shows the dispersions along the cuts crossing
FIG. 8: Detailed band dispersions of sample #2 with 4.3° stacking angle. (a) Constant energy map at $E_F$-100 meV. The Brillouin zone of moiré superlattice is indicated by grey dashed hexagon. (b-d) Band dispersions along the cuts shown in (a). The dispersions from FDCs are indicated by black arrows. (e-g) Band dispersions along the cuts across the ODC and FDCs as shown in (a). (h) MDCs at $E_F$-100 meV for (e-g). The momentum positions of first-generation Dirac points are indicated by black arrows.

The ODC and FDCs simultaneously. The momentum separation between ODC and FDCs is estimated to be $0.23 \pm 0.04 \, \text{Å}^{-1}$ from the MDCs at -100 meV (Fig. 8(h)), consistent with the moiré pattern period extracted from AFM measurements. The ODC and FDCs intersect at around -840 meV, slightly deeper than the -770 meV calculated by moiré pattern period. The difference is possibly attributed to the error bar from AFM measurement. Similar to the graphene/h-BN sample #1 with stacking angle of 2.4°, no obvious signals of SDCs are observed from the constant energy maps or band dispersions.
IV. CONCLUSIONS

We report direct experimental results on the modulated band structure in transferred graphene/h-BN heterostructures with stacking angles of 2.4° and 4.3° respectively. We observed replicas of Dirac cones translated by the reciprocal lattice vectors of the Moiré superlattice from the graphene K point. With variable stacking angle between graphene and h-BN, the size and relative angle of SBZ can be tuned. Unlike previous studies on PE-CVD grown graphene/h-BN samples with 0° stacking angle, no obvious signatures of SDCs are observed from the modulated band structure. This is possibly due to weaker interaction between graphene and h-BN at large stacking angle. It has been reported that at small stacking angle, there are regions of commensurate graphene stretched to fit the lattice of h-BN substrate, and there is a crossover from commensurate to incommensurate states between $\phi = 0$ and $\phi \approx 1.5^\circ$ [15]. Such commensurate to incommensurate transition may also explain the large variations of gaps measured on graphene samples with different stacking angles. Another possibility is that PE-CVD graphene/h-BN samples may have stronger interaction between graphene and h-BN with 0° stacking angle compared to transferred graphene/h-BN samples with tunable stacking angles. More experiments to reveal the atomic structure at the interface with different stacking angles and their correlation with the electronic structures are important to further understand the difference.

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