Recent progress in high pressure X-ray absorption spectroscopy studies at the ODE beamline

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Abstract

High pressure energy-dispersive X-ray absorption spectroscopy is a valuable structural technique, especially when combined with a nano-polycrystalline diamond anvil cell. Here we present recent results obtained using the dispersive setup of the ODE beamline at SOLEIL synchrotron. The effect of pressure and temperature on the X-ray induced photoreduction is discussed on the example of nanocrystalline CuO. The possibility to follow local environment changes during pressure-induced phase transitions is demonstrated for \( \alpha \)-MoO\(_3\) based on the reverse Monte Carlo simulations.

1. Introduction

An alternative to the traditional pattern of beamlines dedicated to the X-ray absorption spectroscopy, based on a two-crystal scanning mode, is given by the use of dispersive optics [1–4]. In this technique, a bent crystal is used as a polychromator, and the setup is usually called Energy Dispersive EXAFS (EDE). The continuous change of the Bragg incidence along the bent crystal opens an energy range in the diffracted beam. The correlation between position and energy of the X-ray is exploited thanks to a position sensitive detector. The setup can be used to acquire both the X-ray absorption near edge structure (XANES) as well as extended X-ray absorption fine structure (EXAFS) [5–7].

The EDE beamline at SOLEIL synchrotron, called ODE for Optic Dispersive EXAFS [8], has extensively developed high pressure XAS (X-ray Absorption Spectroscopy) and XMCD (X-ray Magnetic Circular Dichroism) techniques. The XMCD activity was already presented in [9]. Here we report on some high pressure measurements done with a nano-polycrystalline diamond anvil cell (NDAC) [10].

The main advantages of dispersive XAS are the focusing optics, the short acquisition time and the great stability during the measurements due to the absence of any mechanical movement. These advantages allow the study of small samples, mandatory in the case of high pressure experiments, where a hole in a gasket sandwiched between two diamonds plays the role of the sample chamber. However, the use of single crystal diamonds results in that X-ray absorption spectra become distorted by the presence of the diamond

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Bragg reflections (‘glitches’). The EDE technique is particularly adapted for their reduction (‘deglitching’) due to its real-time visualization of the complete X-ray absorption spectrum that allows the diamonds of a high pressure cell to be oriented in such a way to eliminate glitches. This advantage has disappeared with the advent of nano-polycrystalline diamonds. Their use significantly improves the quality of the experimental signal, eliminating the side effects from glitches [11] and opens the way for many new possibilities, as can be seen from the following contributions of XAS pressure measurements.

The weakness of the ODE beamline is the limited energy range when measuring the EXAFS spectrum. Its main drawback is the difficulty of obtaining the EXAFS spectrum free of parasitic modulations introduced by defects in the optics, which imposes to have very homogeneous samples or to move them randomly to average the effects of texture.

2. X-ray absorption spectroscopy at high pressure

2.1. XANES as a probe of X-ray photoreduction

High-intensity X-ray beams at modern sources of synchrotron radiation can cause radiation damage to a material. The effect is well known in the case of metal–organic complexes [12–15]. This problem may be of particular importance for beamlines having X-ray beam focused to the spot of several tens of microns or less on the sample located in the diamond anvil cell. In particular, when high pressure measurements are performed, the indirect damage of the sample placed in a liquid, which plays the role of the pressure-transmitting medium, can occur due to the radiolysis of the latter. In the radiolysis process, reducing radicals such as solvated electrons and hydrogen atoms are produced upon X-ray irradiation of the solution and are responsible for the sample change [16,17]. For example, the radiolysis can lead to a reduction of metal ions in aqueous solutions and is used for the synthesis of metal nanoparticles [18–20].

Recently, we have demonstrated X-ray induced photoreduction of nanocrystalline CuO, and its dependence on crystallite size, temperature and pressure [21]. Nanocrystalline CuO (nano-CuO) powder samples with the average crystallite size of 8 nm and 20 nm were prepared by a decomposition of Cu(OH)$_2$ precipitate in air at 130°C and 150°C, respectively. The Cu K-edge XANES spectra were collected in the pressure range of 0–23 GPa and the temperature range of 10–300 K. The sample pressure and temperature were controlled using a membrane-type NDAC [10,11] and liquid helium cryostat. The use of NDAC allowed us to accumulate the experimental data free from Bragg peaks due to the diamonds. The polychromatic photon flux on the sample was about $10^9$ photons/s/eV in 25 × 35 μm FWHM. The silicon oil (Rhodorsil Oils 47V100) was used as a pressure-transmitting media.

The sensitivity of the Cu K-edge XANES in nano-CuO (8 nm) to temperature and pressure is shown in Figure 1(a). The emergence of metallic copper is visible at $T = 260$ K and $P = 2$ GPa after 70 min of exposure to X-rays as an increasing shoulder at 8983 eV and a decreasing main peak at 9000 eV. X-ray induced photoreduction occurs more rapidly at higher temperatures and lower pressure. For larger crystallite size (20 nm), the reduction process takes longer time (Figure 1b). At $T = 300$ K and $P = 0.2$ GPa, copper oxide is fully converted into metallic copper after about 90 min of irradiation.
However, an increase of pressure to 23 GPa stabilizes the oxide phase. Note that the photoreduction was not observed for microcrystalline CuO at similar conditions. Thus the rate of nano-CuO photoreduction to metallic copper increases with decreasing nanoparticle size but can be reduced by a decrease of temperature or an increase of pressure. Performing experiment at low temperatures decreases the mobility of reducing radicals [22,23], whereas an increase of pressure results in CuO nanoparticles agglomeration thus restricting their free surface and impeding reduction. The obtained results suggest that possible radiation damage should be taken into account in experiments with high-flux X-ray beams, especially, in the case of nanosized materials.

2.2. Pressure-induced phase transitions probed by EXAFS

The sensitivity of EXAFS to the local atomic structure of a material and recent developments in the EXAFS data analysis based on the reverse Monte Carlo (RMC) simulations [24] provide an invaluable tool to follow pressure-induced phase transitions. However, the availability of high-quality experimental data often remains the main limiting factor to unleash the potential of the RMC method.

The pressure-induced (up to 36 GPa) transformations in 2D layered oxide $\alpha$-MoO$_3$ were studied at the Mo K-edge in [25]. Good quality experimental EXAFS data were acquired using the NDAC cell and allowed us to perform analysis up to 6 Å using the RMC method. The structural models obtained by RMC give the Mo K-edge EXAFS spectra in good agreement with the experimental ones (Figure 2). The corresponding atomic coordinates were used to calculate the radial distribution functions (RDFs) $g(R)$ for Mo–O and Mo–Mo atom pairs as a function of pressure and allowed us to follow the details of local structure transformations upon phase transitions (Figure 2).
At room temperature, molybdenum trioxide exhibits two phase transitions upon compression in the range of 0–43 GPa [26]. \( \alpha \)-MoO\(_3\) transforms to monoclinic MoO\(_3\)-II phase (\( P2_1/m \)) at \( \sim 12 \) GPa, and, next, to monoclinic MoO\(_3\)-III phase (\( P2_1/c \)) at \( \sim 25 \) GPa. The first two phases (\( \alpha \)-MoO\(_3\) and MoO\(_3\)-II) have layered structure composed of strongly distorted MoO\(_6\) octahedra, whereas a collapse of the interlayer gap occurs in MoO\(_3\)-III phase [26].

The difference between \( \alpha \)-MoO\(_3\) and MoO\(_3\)-III phases is well established by EXAFS. The transition is accompanied by a change of the Mo–O–Mo angles between neighbouring molybdenum-oxygen octahedra from \( \sim 142^\circ \) and \( \sim 168^\circ \) in \( \alpha \)-MoO\(_3\) [27] to \( \sim 149^\circ \) in MoO\(_3\)-III [26]. The disappearance of the Mo–O–Mo angle equal to 168° is responsible for a decrease of the second shell peak (at \( \sim 3.2 \) Å in Figure 2) amplitude in the Fourier transform of the EXAFS spectrum of MoO\(_3\)-III due to a reduction of the multiple-scattering (MS) effects within the Mo–O–Mo atomic chains.

The reconstruction of the local environment around molybdenum using the RMC method [24] shed light on the pressure dependence of the RDFs for Mo–O and Mo–Mo atom pairs in details.

At small pressure of about 0.2 GPa, there are three groups of two oxygen atoms each located at \( \sim 1.70, 1.96 \) and 2.26 Å, which form strongly distorted MoO\(_6\) octahedra in \( \alpha \)-MoO\(_3\). At high pressure (35.6 GPa), the collapse of layered structure leads to an increase of molybdenum coordination. Six nearest oxygen atoms from the same layer are responsible for the peaks at \( \sim 1.68, 1.88 \) and 2.12 Å, whereas the seventh oxygen atom bridging two layers is located at \( \sim 2.48 \) Å. The high pressure influences also the

![Figure 2](image-url)
Mo–Mo distribution, leading to a shortening of the distance between neighbouring molybdenum–oxygen polyhedra connected by edges (peaks at 2.9 and 3.25 Å). Thus the change of the molybdenum local environment upon phase transitions is well evidenced by the RMC analysis of high pressure Mo K-edge EXAFS spectra.

3. Conclusion

In this paper, we describe the potential of high pressure energy-dispersive X-ray absorption spectroscopy (XANES/EXAFS) in combination with a nano-polycrystalline diamond anvil cell. The experimental setup is well suited for such measurements, making it possible to obtain experimental data of good quality, being free from the Bragg contribution from diamonds. The accessible spectral range is limited by the size, the quality and the focusing properties of the polychromator crystal at high energy and by the divergence of the incoming beam at low energy. In addition, high intensity of X-rays focused on a sample can cause radiation damage.

Two examples of high pressure XAS studies of transition metal oxide compounds (CuO and MoO₃) are discussed. They illustrate the sensitivity of copper oxide nanoparticles to reduction under X-ray irradiation and the ability to track local structural changes during phase transition in a 2D layered-type molybdenum trioxide using an advanced analysis method based on the reverse Monte Carlo algorithm.

Disclosure statement

No potential conflict of interest was reported by the authors.

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