Communication

Design of Ionic Liquid Crystals Forming Normal-Type Bicontinuous Cubic Phases with a 3D Continuous Ion Conductive Pathway

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Abstract: We have prepared a series of pyridinium-based gemini amphiphiles. They exhibit thermotropic liquid–crystalline behavior depending on their alkyl chain lengths and anion species. By adjusting the alkyl chain lengths and selecting suitable anions, we have obtained an ionic amphiphile that exhibits a normal-type bicontinuous cubic phase from 38 °C to 12 °C on cooling from an isotropic phase. In the bicontinuous cubic liquid–crystalline assembly, the pyridinium-based ionic parts align along a gyroid minimal surface forming a 3D continuous ionic domain while their ionophobic alkyl chains form 3D branched nanochannel networks. This ionic compound can form homogeneous mixtures with a lithium salt and the resultant mixtures keep the ability to form normal-type bicontinuous cubic phases. Ion conduction measurements have been performed for the mixtures on cooling. It has been revealed that the formation of the 3D branched ionophobic nanochannels does not disturb the ion conduction behavior in the ionic domain while it results in the conversion of the state of the mixtures from fluidic liquids to quasi-solids, namely highly viscous liquid crystals. Although the ionic conductivity of the mixtures is in the order of 10⁻⁷ S cm⁻¹ at 40 °C, which is far lower than the values for practical use, the present material design has a potential to pave the way for developing advanced solid electrolytes consisting of two task-specific nanosegregated domains: One is an ionic liquid nano-domain with a 3D continuity for high ionic conductivity and the other is ionophobic nanochannel network domains for high mechanical strength.

Keywords: ionic liquid crystal; bicontinuous cubic phase; gyroid; gemini; ion conduction

1. Introduction

Ionic liquids are a class of organic salts forming liquid states at ambient temperature [1,2]. Owing to the strong electrostatic interactions between the component ions, these ionic liquids have several unique properties, such as negligible volatility, flame retardancy, and high ionic conductivity, that are totally different from those of conventional organic solvents. These properties lead scientists to expect that ionic liquids should be new electrolytes for batteries [3,4], solar cells [5], and fuel cells [6], and therefore intensive efforts have been paid for the design of ion conductive materials based on ionic liquids [7]. One of a unique approach for enhancing their ionic conductivities is to align ionic liquids into ordered states [8]. For example, Kato and Ohno succeeded in aligning ionic liquids into 1D [9,10], 2D [11,12], and 3D [13,14] manners by endowing ionic liquids with liquid crystallinity. For the design of liquid-crystalline (LC) ionic liquids (namely ionic liquid crystals), it is important
not only to design the shape of molecules and inter/intra-molecular interactions but also to control the volume balance between the ionic and non-ionic parts. The relationships between the molecular structure and molecular assembled structures of ionic liquid crystals are well-summarized in several reviews and papers [8,15–18].

In 2007, Kato and Ichikawa reported that a series of ammonium-based ionic liquid crystals self-organize into bicontinuous cubic (Cub_{bi}) LC phases with three-dimensionally continuous ionic nanochannel networks (Figure 1a) [13]. This phase can be categorized into an “inverted-type” Cub_{bi} phase. This is a situation where ionic nanochannels (Figure 1a, red) expand three-dimensionally in an ionophobic alkyl chain domain (Figure 1a, blue). In addition, it was revealed that the ionic nanochannels function as ion transporting pathways because the 3D branched nanochannel structure enables ions to transport over LC domain boundaries.

In the course of studies on nanosegregated liquid crystals, it has been well-known that there are “normal-type” nanosegregated LC phases whose nanostructures are the reverse of those of inverted-type LC phases [19,20]. This is applicable in the case of Cub_{bi} phases [21–25]. This well-known fact stimulates us to have an interest in the potential utility of normal-type Cub_{bi} phases (Figure 1c) as ion conductive materials. Here, we have designed an ionic liquid crystal forming a normal-type Cub_{bi} phase and have then evaluated its potential utility as an ion conductive material.

**Figure 1.** (a) A series of wedge-shaped ammonium salts forming inverted-type bicontinuous cubic (Cub_{bi}) phases [13]. (b) Molecular structure of gemini-type amphiphile 1–4. (c) A schematic illustration of a normal-type Cub_{bi} phase.

### 2. Results and Discussion

For the design of ionic liquid crystals forming normal-type Cub_{bi} phases, we focus here on the use of gemini-type amphiphilic structures. One of the reasons that we focus on the gemini design is that there have been a relatively large number of reports on gemini-type amphiphiles exhibiting lyotropic Cub_{bi} phases in the presence of water [26–29]. Moreover, just recently, we have succeeded in developing gemini-type ionic amphiphiles exhibiting thermotropic Cub_{bi} phases [30,31]. Based on the molecular design in these previous studies, we have designed a series of gemini-type ionic amphiphiles, 1–4, with various alkyl chain lengths and anion species (Figure 1b). The number of carbon atoms in the alkyl chains is 14 or 16. As anion species, tetrafluoroborate (BF_{4}), trifluoromethanesulfonate (Tf), and bis(trifluoromethane)sulfonimide (Tf_{2}N) have been selected because they are known as useful anions for designing ionic liquids with a low glass transition temperature (T_g) [32]. As a cation part, a pyridinium cation has been employed since there have been several reports on the exhibition of...
Cubbi phases for pyridinium-based amphiphiles [33–37]. As a linker, we have selected a hexyl chain according to several researches on gemini-type amphiphiles [26–31].

The phase transition behavior of amphiphiles 1–4 was examined using a polarizing optical microscope (POM), differential scanning calorimetry (DSC), and X-ray diffraction (XRD) measurements. The obtained results are summarized in Table 1. It has been found that 1 shows no birefringence (Figure 2a) while two distinct XRD peaks are found at 2.44° and 2.80° at 30 °C (Figure 2b). The reciprocal ratio of these d-spacing values is $\sqrt{6}:\sqrt{8}$ and therefore these peaks can be indexed as the reflection of (211) and (220) in a Cubbi phase with $Ia3d$ symmetry. Based on DSC thermograms, we have revealed that 1 forms a Cubbi phase from 38 °C to 12 °C on cooling (Figure 2c). Nanosegregation of the ionic and non-ionic parts is a major driving force for the spontaneous molecular organization. It is of interest that the selection of the anion species plays a critical role for controlling their self-organization behavior. For example, the substitution of BF$_4$ anion with Tf anion causes a loss to its liquid crystallinity. Compound 2 shows no mesomorphic behavior. Considering the difference between the anion radii of these two anions, it is assumed that the strength of electrostatic interactions between the cation and anion is one of the governing factors that influences the spontaneous progress of nanosegregation of the ionic parts and non-ionic parts. Compound 3 has longer alkyl chains than those of 1, yet also exhibits LC behavior while the observed LC phase is not a Cubbi but a Sm phase. The characterization of the Sm phase was carried out in the same way (Figure 2d–f). The disappearance of LC properties is also induced by substituting a BF$_4$ anion with a Tf$_2$N anion with a larger anion radius. Considering that the results of compounds 1 and 3 showed that the elongation of the alkyl chains leads to the change of the mesophase pattern from a Cubbi phase with a curved ionic/non-ionic interface to a smectic A (SmA) phase with flat interfaces, it is presumed that the Cubbi phase is not an inverted-type Cubbi phase, but a normal-type Cubbi phase consisting of an ionic sheath domain and 3D branched ionophobic domains (Figure 1c).

Table 1. Phase transition behavior of compound 1–5.

| Compound | Phase Transition Behavior a) |
|----------|-----------------------------|
| 1        | Iso 38 (-) Cubbi 12 (-14) Cr |
| 2        | Iso -8 (-20) Cr             |
| 3        | Iso 109 (-0.4) SmA 33 (-39) Cr' |
| 4        | Iso -3 (-27) Cr            |
| 5        | Iso 10 (-38) Cr            |

a) Transition temperatures (°C) and enthalpies of transition (mJ/mg, in parentheses) determined by DSC thermograms on the cooling rates of 10 °C min$^{-1}$. Cr, Cr': Crystalline; Cubbi: Normal-type bicontinuous cubic; SmA: Smectic A; Iso: Isotropic.
Comparing the phase behavior of 1 and 5, it has been clearly found that the crystallinity of Compound 5 shows no mesomorphic behavior, which is confirmed by the DSC measurement. Here, we define the molar fraction of LiBF₄ (X). For example, the dissolution of the two components into a methanol, and subsequent slow evaporation of the solvent.

Homogeneous mixtures of these two components (1/LiBF₄) in various molar ratios were prepared by dissolving the two components into a methanol, and subsequent slow evaporation of the solvent. We assume that the connection of the two amphiphilic molecules causes two effects. One is that the connection makes it difficult for their alkyl chains to get a suitable position for dense packing, which leads to the lowering of the crystallinity. The other is that the connection enhances the stability of the nanosegregated states, which strengthens the self-organization ability of these amphiphiles.

Compatibility of 1 and a lithium salt, lithium tetrafluoroborate (LiBF₄), was examined. Homogeneous mixtures of these two components (1/LiBF₄) in various molar ratios were prepared by dissolving the two components into a methanol, and subsequent slow evaporation of the solvent. Here, we define the molar fraction of LiBF₄ in the 1/LiBF₄ mixtures as X. By observing the obtained mixtures under POM, we found that these two components form homogeneous states where LiBF₄ is dissolved into the ionic domains in the nanosegregated states. It is of interest that the 1/LiBF₄...
mixtures exhibit not only Cub_ bi phases but also Col phases depending on the temperature and the molar fraction of LiBF_4 (X). For example, the 1/LiBF_4 (X = 0.09) mixture exhibits a Col phase from 44 °C and then turns into a Cub_ bi phase at 40 °C on cooling. The phase transition from the Col to Cub_ bi phase was definitively confirmed by POM observation. Upon cooling from an isotropic liquid state, an appearance of a focal conic fan texture is observed at 44 °C (Figure 4). On further cooling, black domains with non-birefringent domains appear at various positions at 40 °C and they grow until the birefringence disappears entirely. On the other hand, the 1/LiBF_4 (X = 0.20) mixture exhibits only a Col phase. The obtained results are summarized as a phase diagram (Figure 5). It can be seen that the addition of LiBF_4 induces and stabilizes the exhibition of Col phases. We attribute the exhibition of the Col phases to the increase of the volume ratio of the ionic domains against the non-ionic domains.

**Figure 4.** Polarizing optical microscopic images of the 1/LiBF_4 (X = 0.09) mixture on the phase transition from the Col to the Cub_ bi phase.

**Figure 5.** Phase diagram of the 1/LiBF_4 mixtures as a function of the mole fraction of LiBF_4 (X) in the mixtures. Cr, crystalline; Col, columnar; Cub_ bi, bicontinuous cubic; Iso, isotropic.

Ionic conductivity (σ) of 1 and the 1/LiBF_4 mixtures was examined using alternating current impedance method with a Schlumberger Solartron 1260 impedance analyzer on cooling at a rate of 1 °C min⁻¹ (frequency range: 10 Hz–10 MHz, applied voltage: 0.3 V). The measurements were carried out upon cooling from 100 °C to room temperature. The ionic conductivity on each temperature was estimated from a Cole–Cole plot (Figure 6a). The estimated σ values are plotted against the reciprocal absolute temperature. It has been found that 1 shows σ of 1.6 × 10⁻⁶ S cm⁻¹ at 70 °C, which linearly decreases upon cooling. In general, the ionic conductivity of ionic liquid crystals drastically decreases upon the phase transition from isotropic liquid states to polydomain LC phases [9] for of two reasons. One is the drastic increase of the viscosity of the materials, which largely lowers the diffusion constants of the component ions. The other is the formation of domain boundaries that disturb the macroscopic
ion migration. It should be noted that the latter can be overcome when the LC domains are aligned in a macroscopic scale [9]. In the case of compound 1, such drastic decrease of ionic conductivity is not found at 38 °C where 1 shows the phase transition from the isotropic liquid to the Cubbi LC phase with high viscosity, although no alignment control of the LC domains is performed. These results lead us to conclude that the formation of the 3D branched ionophobic nanochannels in the ionic domain does not disturb the ion conduction. Namely, the 3D ionophobic channels endow the ionic domain with a quasi-solidity while it does not act as an obstacle that disturbs the macroscopic ion conduction in the ionic domain. This effect is similar to those that have been found for gelation of ionic liquids by some polymers [38] and supramolecular fibers [39].

The ionic conductivities of the 1/LiBF$_4$ ($X = 0.09$ and $0.33$) mixtures were also measured in the same way. The obtained results are compared in Figure 6b. It has been found that the ionic conductivities of the 1/LiBF$_4$ ($X = 0.09$) mixture are slightly higher than those of pristine 1 while the further addition of LiBF$_4$ results in the lowering of ionic conductivities. These results can be explained by the competition of two effects: one is the increase of ion carrier densities in the mixture and the other is the increase of the viscosity that is induced by the presence of the lithium cation that is harder than the pyridinium cation of 1.

In the decades of studies on ionic liquids, a variety of potential utilities of ionic liquids have been explored for solvents of biopolymers [40,41], gas treatments [42,43], and drug delivery [44]. We expect that organized ionic liquids, including the present systems, will be useful not only for the development of electrolytes but also for these new fields.

![Figure 6](image_url)

**Figure 6.** (a) A Cole–Cole plot for compound 1 at 36 °C in the Cubbi phase. (b) Arrhenius plots of ionic conductivities of compound 1 and the 1/LiBF$_4$ mixtures ($X = 0.09$ and $0.33$).

3. Materials and Methods

**Thermotropic Liquid–Crystalline Properties**

Thermotropic liquid–crystalline behaviors of compounds 1–5 and the 1/LiBF$_4$ mixtures were examined by polarizing optical microscope (POM, Olympus BX51) and X-ray diffraction (XRD, Rigaku SmartLab X-Ray Diffractometer) measurements. Some obtained results are shown in the Supplementary Materials. Differential scanning calorimetry (DSC, Hitachi DSC7000X) measurements were also performed for the mixtures.

4. Conclusions

In conclusion, we have succeeded in designing gemini-type ionic liquid crystals that form normal-type bicontinuous cubic and smectic A phases. For the induction of normal-type bicontinuous cubic phases, the selection of anion species and the employment of long alkyl chains with suitable lengths are found to be important. The normal-type bicontinuous cubic phase consists of two continuous nanosegregated domains: one is a continuous ionic domain and the other is 3D branched ionophobic
nanochannel domains that spread around the ionic domain. These liquid-crystalline materials can dissolve lithium salts as well as conventional ionic liquids. The dissolution of lithium salts influences their mesophase behavior. The ionic conductivities of these liquid–crystalline materials are in the order of $10^{-7}$ S cm$^{-1}$ at $36 \, ^\circ{C}$. Although these values are lower than those of conventional ionic liquids, we believe the present material design will be one of the potential approaches for creating a quasi-solid electrolyte consisting of two task-specific domains: One is a 3D continuous ionic domain for high ionic conductivity and the other is 3D continuous ionophobic nanochannels that contribute to endowing the matrix with high mechanical strength without disturbing the ion conduction in the neighbouring ionic domain.

**Supplementary Materials:** The following are available online at http://www.mdpi.com/2073-4352/9/6/309/s1.

**Author Contributions:** T.I., Y.S., T.K., H.O. (Hikaru Oshiro), A.O., and H.O. (Hiroyuki Ohno) conceived and designed the experiments; Y.S., T.K., H.O. (Hikaru Oshiro), and A.O performed the experiments and analyzed the data; Y.S., T.K., H.O. (Hiroyuki Oshiro), A.O., and T.I. wrote the paper.

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**Conflicts of Interest:** The authors declare no conflicts of interest.

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