A preliminary attempt has been made by Bhattacharya et al.\textsuperscript{28} to predict the oxidation kinetics of complex multi-component alloys by ML. Approach has several advantages in handling complex multi-component alloys and offers the potential to extract insights from complex experiments or synthetic datasets generated from physics-based simulations. Thus, modern data analytics approaches can be considered an alternate and/or complementary tool to accelerate the design and development of materials.

Although of practical importance, there is no established modeling framework to accurately predict high-temperature cyclic oxidation kinetics of multi-component alloys due to the inherent complexity. We present a data analytics approach to predict the oxidation rate constant of NiCr-based alloys as a function of composition and temperature with a highly consistent and well-curated experimental dataset. Two characteristic oxidation models, i.e., a simple parabolic law and a statistical cyclic oxidation model, have been chosen to numerically represent the high-temperature oxidation kinetics of commercial and model NiCr-based alloys. We have successfully trained machine learning (ML) models using highly ranked key input features identified by correlation analysis to accurately predict experimental parabolic rate constants ($k_p$). This study demonstrates the potential of ML approaches to predict oxidation kinetics of alloys over wide composition and temperature ranges. This approach can also serve as a basis for introducing more physically meaningful ML input features to predict the comprehensive cyclic oxidation behavior of multi-component high-temperature alloys with proper constraints based on the known underlying mechanisms.

### RESULTS AND DISCUSSION

**Numerical representation of cyclic oxidation data**

Extensive experimental investigations of the oxidation kinetics of 25 model NiCr alloys and four commercial Ni-based alloys (i.e., Nimonic 80 A, Nimonic 90, René 41, and Haynes 282) have been performed in both dry air and wet air (air + 10% water vapor) atmospheres at Oak Ridge National Laboratory (ORNL). Figure 2 shows examples of the cyclic oxidation behavior of select NiCr alloys\textsuperscript{37,38} in the dataset. The alloy composition ranges, experimental conditions, and details of the dataset are available in the Supporting Information.
summarized in Table 1. This consistently measured oxidation and well-curated dataset, collected over years of experiments at ORNL, enabled exploring the potential of ML to predict the oxidation kinetics of multi-component high-temperature alloys. However, raw cyclic oxidation data, which is given as a $2 \times N$ matrix (cycle vs. mass change) per alloy, cannot be directly used in ML because singular numeric values are needed as the target for the regression-type ML training. Hence, it is necessary to find a proper numerical representation for the oxidation kinetics of each alloy so that the correlation between input features and $k_p$ can be evaluated.

High-temperature oxidation of multi-component alloys is a complex dynamic process, which requires concurrent consideration of both thermodynamics and diffusion kinetics. As shown in Fig. 2, the addition of certain minor alloying elements can significantly alter the mass change behavior of some alloys during oxidation. Thus, identifying a proper numerical representation for such complex phenomena is the key to building reliable ML models to predict the high-temperature oxidation kinetics of NiCr-based alloys. Mathematically, any mass change curve can be fitted with an arbitrary polynomial. However, parameters obtained from such regression without any physical meaning will not allow gaining insights into the oxidation mechanisms and compare alloy oxidation performance. Thus, we have focused on identifying widely accepted oxidation models with a robust formalism to determine physically meaningful parameters to represent the oxidation kinetics of the studied alloys.

As illustrated in Fig. 3, two oxidation models, i.e., a simple parabolic law ($\Delta m = \sqrt{k_p t}$, where $\Delta m$ is the mass gain and $t$ is the oxidation time; $s$-$k_p$ hereafter) and a statistical cyclic oxidation model ($p$-$k_p$ hereafter) were adopted in this work. Both models have been widely used to interpret the oxidation kinetics of numerous alloys. In this study, the $s$-$k_p$ model evaluated the mass change data up to 100 h to exclusively represent growth rates with the minimized influence of mass loss. The $p$-$k_p$ model, consisting of two parameters, i.e., $p$ (discrete oxide spallation probability) and $k_p$, evaluated the complete curve, thus, enabling us to assess the capability of the ML approach to predict oxidation kinetics involving both oxide growth and loss processes. It is anticipated that the derived $k_p$ will be a good numerical representation of our raw experimental cyclic oxidation data and can serve as proper target property for analyzing the oxidation kinetics.
kinetics by the data analytics approach. Since the oxidation mechanisms of NiCr-based alloys in dry and wet air are different\textsuperscript{49,50}, the complete experimental data after being fitted by each oxidation model were divided into two datasets based on the oxidation atmosphere. Finally, four datasets, \(s-k_p\) and \(p-k_p\), in dry and wet air were generated from the existing experimental data for further analysis in this study (see Table 1).

**Correlation analyses**

We have used MIC and PCC methods to quantitatively analyze the correlation between all input features (alloy compositions and oxidation temperature) and \(k_p\) respectively. While PCC can evaluate the strength of the positive and negative, but only linear relationships between two variables. The MIC method can identify the strength of both linear and nonlinear relationships, but only with a sign. Both approaches are expected to provide insights into the correlation between input features and \(k_p\) from differing statistical aspects, which may inspire alloy design experts to generate alloy hypotheses\textsuperscript{23,26}. Moreover, correlation analysis can facilitate the training of high-fidelity ML models using highly ranked features. For the PCC analysis, a positive correlation between a feature and \(k_p\) here implies that the increase of the feature will increase \(k_p\), i.e., a higher oxidation rate, and vice versa. Figure 4 shows the PCC analysis results for the \(p-k_p\) of dry and wet air datasets as examples. The remaining correlation analysis results are presented in Supplementary Figs. 1 and 2. It is worth mentioning that although as the base element, Ni content is the balance of remaining alloying elements, we still use Ni content as an input feature since ML algorithms are only solving a mathematical problem and do not know such a correlation. The oxidation temperature (T) was identified as the feature having the most substantial impact on \(k_p\) in all cases. It is encouraging that correlation analysis replicated the community’s existing knowledge: oxidation temperature is typically the factor that most strongly affects oxidation rate. PCC also correctly identified that temperature has the most positive correlation with \(k_p\), consistent with the fact that the oxidation rate increases with increasing temperature.

It can also be observed in Fig. 4 and Supplementary Fig. 1 that chromium (Cr) as the major alloying element in the studied alloys was identified to have the most negative correlation with \(k_p\), which is consistent with the fact that the Cr addition is beneficial to the formation of an external chromia solid-state diffusion barrier that slows the oxidation reaction as it increases in thickness\textsuperscript{51,52}. Most of the alloys in this dataset were designed to form protective chromia scales. Aluminum (Al) is expected to further decelerate the growth rate of the chromia scale for NiCr-based alloys\textsuperscript{53}; thus, the negative coefficient of Al in wet air (Fig. 4b) is reasonable. Fe is known to form fast-growing oxides\textsuperscript{54,55}. Its second strongest positive correlations with \(k_p\) in both dry and wet air (Fig. 4) are in accordance with this understanding. Similar correlations were also observed for Cr.
evaporation. Although our dataset was collected over a number of oxidation conditions to fully facilitate data analytics for the classes of alloys included. Certainly, this limitation could most likely be said of any existing materials dataset, which includes complex alloy compositions, structures, and behavior. However, the correctly identified correlation in Fig. 4, Supplementary Figs. 1 and 2 still clearly demonstrated the value of correlation analysis in applying data analytics to materials science. The intention of performing correlation analysis here is not only to gain insights into the impact of input features on \( k_p \) but also to have a numerical basis for the selection of input features to train ML models for understanding their influence on the performance of ML models.

**Machine learning**

Based on the ranking of features from correlation analyses, five widely used ML models, i.e., LR, BR, NN, RF, and SVM, have been trained using different numbers of top-ranking features to evaluate their performances. For models trained with the dry and wet air \( p-k_p \) datasets, Fig. 5 presents their average performance, represented by the Nash and Sutcliffe coefficient of efficiency (NSE)\(^{36}\), and corresponding standard deviation as a function of the numbers of top-ranking features based on the rankings of PCC (the absolute values of PCC in Fig. 4). The remaining results are presented in Supplementary Figs. 3 and 4. Since NSE alone could not fully qualify the performance of these models, another metric, i.e., coefficient of determination \( R^2 \) COD, and a band showing the goodness-of-fit were adopted to facilitate the assessment of ML models, as presented in Fig. 6.

NSE here was mainly used to compare the performance of these models. As shown in Fig. 5 and Supplementary Fig. 3, the performance of ML models trained with dry air datasets shows a similar trend to those with wet air datasets. In most cases, increasing the top-ranking features from 2 to \(-4 \) or \(-6 \) increased the NSE of these models markedly. Intriguingly, considering more features did not considerably improve the performance of the obtained models. The NSE of the models trained with the wet air datasets is relatively lower than their dry air datasets counterparts. This trend can be attributed to the increased variation of measured oxidation kinetics of alloys studied in wet air exposure. For the dry air dataset, the maximum NSE of all models is between 0.6 and 0.75. Among the considered models, SVM has the highest NSE (>0.7) in all cases (Fig. 5 and Supplementary Figs. 3 and 4). It reaches the maximum NSE when using the top 3 to 5 features. Afterward, it gradually decreases with an increasing number of features, revealing that in this study including features with a ranking lower than the 5th is not beneficial to the performance of ML models. For the wet air dataset, the maximum NSE of all models is only between 0.45 and 0.60. In general, the top four to six features in this study were required to achieve an NSE of 0.5 or higher. SVM still exhibits the highest accuracy among considered models. As discussed in the previous section, most of these features are experimentally identified features that significantly impact the oxidation of NiCr-based alloys.

In all cases, the performance of these ML models evaluated with NSE is generally in the order of SVM>BR>RF>NN>LR (in dry air) and SVM>BR>LR>NN>RF (in wet air). Results show that these models are sensitive to the number of features considered in training but to different extents. SVM performs similarly to the linear-based models BR and LR, particularly for those trained with the wet air datasets. A linear kernel function (assuming the input features and \( k_p \) is in a linear relationship) was adopted for the present SVM models after hyperparameter tuning. Considering more than the top six to eight features significantly degraded the performance of LR for the dry air dataset, whereas this was not the case for the wet air dataset. This could be caused by the smaller data volume of the dry air dataset than the wet air dataset. BR uses probability distributions to formulate LR, rather than point estimates used by LR and identifies the posterior distribution for the model parameters. Thus, it is believed that BR is more tolerant of overfitting\(^{32,33}\). This is why, in this case, the NSE of the BR model
the dimensionality of data, the accuracy of NN does not decrease
the number of features increases, accompanying the increase of
However, as shown in Fig. 5, Supplementary Figs. 3 and 4, when
becomes less representative in a higher-dimensional space.

because its distance measure, i.e., the Euclidean distance,
points in close to each other and simply outputs the average value of data
accuracy. NN hinges on the assumption that similar things are
Therefore, the performance of RF is not sensitive to the number of
ensemble learning method, RF assigns different importance to
independent of the number of considered features.

does not notably decrease with an increasing number of features. RF in
this study generally requires the inclusion of the top four features to obtain the NSE of >0.65 (for dry air datasets) and >0.5
for wet air datasets; after that, its performance was almost independent of the number of considered features.

The observed trend can be understood as follows: as an
ensemble learning method, RF assigns different importance to
each feature during model training; thus, less critical features
would have less or even no contribution to its performance. Therefore, the performance of RF is not sensitive to the number of features;
considered in training after it reaches the maximum accuracy. NN hinges on the assumption that similar things are
close to each other and simply outputs the average value of data points in k-nearest neighbors. The performance of NN usually
degrades as the dimensionality of data increases significantly, because its distance measure, i.e., the Euclidean distance, becomes less representative in a higher-dimensional space. However, as shown in Fig. 5, Supplementary Figs. 3 and 4, when
the number of features increases, accompanying the increase of the dimensionality of data, the accuracy of NN does not decrease significantly. This finding indicates that the dimensionality of data

in this study does not induce a significant adverse effect on the performance of NN. Since SVM possess the best performance among all considered ML models, its performance does not significantly degrade with increasing numbers of features.

SVM models trained with all 11 features in our datasets are believed to be the most promising ML models for predicting the $k_p$ of NiCr-based alloys since they exhibit relatively good performance and concurrently consider essential features from a high-temperature oxidation perspective. Comparisons between the experimental and predicted $k_p$ by the SVM models trained with all 11 features are presented in Fig. 6 (for $p-k_p$) and Supplementary Fig. 5 (for $s-k_p$). While the accuracy of trained ML models evaluated with NSE is around 0.7, it is encouraging that most of the predicted values are within the acceptable deviation range. In addition, the other metric $R^2$ COD for both dry and wet air ($p-k_p$) is close to 0.9 or higher, which strongly indicates that our ML models can efficiently capture the trend of cyclic oxidation response of NiCr-based alloys as a function of alloy compositions and oxidation temperature.

In practice, the same alloy tested under the same condition may exhibit significant test-to-test variations, particularly in wet air testing. For example, the $k_p$ from the $p-k_p$ model of Nimonic 80 A at 950°C in wet air from six tests varied significantly (Fig. 7). Thus, dashed lines, as the acceptable deviation, representing one order of magnitude difference between the predicted and experimental $k_p$, are superimposed in Fig. 6 and Supplementary Fig. 5. The predicted $k_p$ correctly captures the experimental data trend, and most of the data points lie between the dashed lines. This indicates good agreement between experimental and predicted $k_p$ was achieved from a high-temperature oxidation perspective for the multi-component commercial alloys (group I). This conclusion is further supported by the high $R^2$ COD of ≥0.9 in both cases. Data in Fig. 6b are slightly more scattered than those in Fig. 6a, attributing to the increased spallation probability and thus more considerable test-to-test variation in wet air exposure.

Although a satisfactory agreement between the experimental and predicted $k_p$ was achieved, more work is required in the future. Firstly, the performance of ML models depends not only on what features have been considered but also on the nature and repartition of the dataset used for training; thus, further detailed analysis of this aspect is required. Secondly, $k_p$ alone could not fully capture the complete oxidation kinetics of alloys. Other properties, such as spallation probability and oxidation lifetime, should also be concurrently considered as target properties to be modeled/predicted. Thirdly, besides the alloy composition and temperature considered in this study as input features, the high-temperature oxidation kinetics are also affected by factors like microstructure, oxide scale spallation, alloy surface depletion of
elements that are selectively oxidizing, and oxide evaporation in wet air atmosphere. Features that can well represent these additional factors and ML frameworks that can incorporate these factors as much as possible will be highly conducive to establishing high-fidelity surrogate models.

Additionally, material datasets commonly suffer from the uneven data distribution. While the present dataset can be regarded as one of the largest and the most comprehensive cyclic oxidation dataset for high-temperature NiCr-based alloys, many gaps are identified. A highly desirable alloy dataset to apply data analytics would have an even distribution of data points of all elements without major gaps in a high-dimensional space. The movement toward such an ideal alloy dataset can be obtained by exercising a design of experiments (DOE) campaign to fill key gaps in the existing dataset(s). However, the dataset used in the current study is biased to some extent and has a number of gaps, as shown in Fig. 8. This is attributed to the fact that the objective of past empirical alloy research has been mainly focused on identifying alloys with superior properties over existing ones. Consequently, the alloy composition variation strategy has been heavily geared toward improving properties, more often ending up in biased and narrow boundary conditions from a data analytics perspective. It is evident that alloy data with poor properties can serve as equally useful learning input for ML models; however, an effort to strategically collect such data to fill the gaps has been overlooked, not only by us but likely throughout the materials community. Hence, a strategy that can identify the most important gaps in an existing dataset for alloy data analytics and efficiently augment a small amount of experimental data needs to be developed to leverage legacy data, particularly for complex materials.

In summary, we demonstrated a workflow of applying a data analytics approach to predict the cyclic oxidation kinetics of NiCr-based alloys using a highly consistent and well-curated experimental dataset collected over many years and numerous different studies at ORNL. The oxidation data in dry and wet air as a function of alloy compositions and oxidation temperatures (800–950 °C) were used to train ML models. Identifying a proper numerical representation was the critical procedure for successfully predicting the high-temperature oxidation kinetics of NiCr-based alloys. Oxidation kinetics of alloys was represented by the $k_p$ from properly selected oxidation models, i.e., a simple parabolic growth law for data up to 100 h ($s$-$k_p$) and a statistical cyclic oxidation model for the complete curve ($p$-$k_p$, up to 2000 h), respectively.

Correlation analysis and training of ML models (i.e., BR, LR, NN, RF, and SVM regression) were performed to quantitatively evaluate the inter-relationship between input features and $k_p$. The influence of top-ranking features on the performance of considered ML models was also identified. The trained ML models were able to predict $k_p$ from both $s$-$k_p$ and $p$-$k_p$ models satisfactorily. The typical test-to-test variation in experimental high-temperature oxidation data was believed to be the primary source for the discrepancy between the ML-predicted and experimental $k_p$. The current results demonstrated the potential of the ML approach to predict the oxidation kinetics of multi-component NiCr alloys with wide composition and temperature ranges. We anticipate that this data analytics method can be applied to predict the oxidation kinetics of other classes of high-temperature alloys.

**METHODS**

**Experimental procedure**

All specimens with the dimensions of ~$10 \times 20 \times 1.5$ mm$^3$ were ground to a 600-grit finish and cleaned ultrasonically in acetone and methanol prior to cyclic oxidation experiments. A 1-h-cycle consisting of a 60 min exposure at temperature and 10 min cooling in automated cyclic rigs was adopted in this work. The specimens were exposed at temperatures between 800 and 950 °C in flowing dry and wet air (air + 10% water vapor), respectively, with a gas flow rate of 300 cm$^3$ min$^{-1}$ for up to 2000 h.

**Correlation analysis and machine learning**

The correlation between the input features and $k_p$ was evaluated by PCC, respectively. PCC considers the strength of the linear relationship between two variables, while MIC can identify the strength of both linear and nonlinear relationships. The correlation coefficient of PCC is between −1 and 1, where 1 indicates a total positive linear correlation, −1 represents a complete negative/reciprocal linear correlation, and 0 indicates no correlation. The correlation coefficient of MIC ranges between 0 and 1. Coefficient values close to 1 or −1 indicate the strongest correlation between the variables and target property.

Five representative ML models were used: LR, BR, RF, and SVM regression. A brief introduction to these models is provided below. For details, please refer to the corresponding literature. LR models the relationship between input and output variables by fitting a linear equation. Parameters of the LR model are fitted to minimize the residual sum of squares between input and output data. BR is another LR model. It formulates an LR using probability distributions rather than point estimates that LR does. The output is assumed to be drawn from a probability distribution instead of being estimated as a single value. k-NN outputs the average of the values of given data points in the k-nearest neighbors. Since the function is approximated only locally, it only uses a subset of the relevant dataset. RF is an ensemble learning method that constructs multiple decision trees during training and outputs the mean prediction of the individual trees. SVM can be used for both classification and regression problems. For a classification problem, SVM constructs a set of hyperplanes in high-dimensional space to distinctly classify the data points. For a regression problem, SVM is generalized by introducing an ε-insensitive region (ε-tube) around the function. This tube reformulates the regression problem to find a function exhibiting the least deviation from the obtained targets across all training data while balancing model complexity and prediction error.

These ML models were trained with different numbers of top-ranking features based on the ranking from MIC or PCC (the absolute value of
fold cross-validation approach with the model predicted value. The NSE is calculated as follows:

\[
NSE = 1 - \frac{1}{\sum } (\bar{O}_i - P_i)^2
\]

where \( \overline{O} \) and \( \overline{P} \) are the observed values and their mean, respectively, \( P \) is the model predicted value. \( N \) is the number of data in the dataset. The k-fold cross-validation approach \( 64 \) with \( k = 5 \) was used in ML training. This approach randomly divided the input data into \( k \) groups. One group (i.e., unseen data) was withheld during training, and the remainder \( k \)-1 groups were used to train the ML model. Then the unseen dataset was used to evaluate the accuracy of models.

**DATA AVAILABILITY**

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

**CODE AVAILABILITY**

Codes used in this work are available via GitHub (https://github.com/ornlpmcp/ASCENDS).

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**REFERENCES**

1. Birks, N., Meier, G. H. & Pettit, F. S. Introduction to the High Temperature Oxidation of Metals. (Cambridge Univ. Press, 2006).
2. Young, D. J. High Temperature Oxidation and Corrosion of Metals. 2nd edn (Elsevier, 2016).
3. Das, N. K. & Shoju, T. Early stage oxidation of Ni-Cr binary alloy (111), (110) and (100) surfaces: a combined density functional and quantum chemical molecular dynamics study. Corros. Sci. 73, 18–31 (2013).
4. Ohler, B., Prada, S., Pacchioni, G. & Langel, W. DFT simulations of titanium oxide films on titanium metal. J. Phys. Chem. C 117, 358–367 (2013).
5. Hong, K.-H., Kim, J.-H., Chang, K. & Kwon, J. The role of Cr on oxide formation in Ni-Cr alloys: a theoretical study. Comput. Mater. Sci. 142, 185–191 (2018).
6. Qi, J., Xu, H., Liang, Z., Lu, P. & Zhou, C. The role of Cr atom in the early steam oxidation of Fe-based alloys: an atomistic simulation. Mater. Corros. 72, 465–473 (2020).
7. Tedmon, C. S. The effect of oxide volatilization on the oxidation kinetics of Cr and Fe-Cr Alloys. J. Electrochem. Soc. 113, 766–768 (1966).
8. Vaché, N., Cadoret, Y., Dods, B. & Monceau, D. Modeling the oxidation kinetics of titanium alloys: review, method and application to Ti-64 and Ti-6242s alloys. Corros. Sci. 178, 109041 (2020).
9. Chyrkin, A. et al. Modeling carbide dissolution in alloy 602 CA during high temperature oxidation. Corros. Sci. 96, 32–41 (2015).
10. Pillai, R. et al. External alpha-Al2O3 scale on Ni-base alloy 602 CA - Part II: microstructural evolution. Corros. Sci. 127, 27–38 (2017).
11. Chyrkin, A. et al. Modelling compositional changes in nickel base-allow 602 CA during high temperature oxidation. Mater. High. Temp. 32, 102–112 (2014).
12. Ramprasad, R., Batra, R., Pilania, G., Mannodi-Kanakkithodi, A. & Kim, C. Machine learning in materials informatics: recent applications and prospects. npj Comput. Mater. 3, 54 (2017).
13. Schmidt, J., Marques, M. R., Botti, S. & Marques, M. A. Recent advances and applications of machine learning in solid-state materials science. npj Comput. Mater. 5, 1–36 (2019).
14. Ramakrishna, S. et al. Materials informatics. J. Intell. Manuf. 30, 2307–2326 (2019).
15. Bock, F. E. et al. A review of the application of machine learning and data mining approaches in continuum materials mechanics. Front. Mater. 6, 110 (2019).
16. Mereidig, B. et al. Combinatorial screening for new materials in unconstrained composition space with machine learning. Phys. Rev. B 89, 094104 (2014).
17. Pilania, G., Wang, C., Jiang, X., Rajasekaran, S. & Ramprasad, R. Accelerating materials property predictions using machine learning. Sci. Rep. 3, 2810 (2013).
18. Pei, Z., Yin, J., Hawk, J. A., Alman, D. E. & Gao, M. C. Machine-learning informed prediction of high-entropy solid solution formation: beyond the Hume-Rothery rules. npj Comput. Mater. 6, 1–8 (2020).
19. Lee, S., Peng, J., Shin, D. & Choi, Y. S. Data analytics approach for melt-pool geometries in metal additive manufacturing. Sci. Technol. Adv. Mater. 20, 972–978 (2019).
20. Alberi, K. et al. The 2019 materials by design roadmap. J. Phys. D: Appl. Phys. 52, 013001 (2018).
21. Rajan, K. Materials informatics. Mater. Today 8, 38–45 (2005).
22. Peng, J. et al. Uncertainty quantification of machine learning predicted creep property of alumina-forming austenitic steels. JOM 73, 164–173 (2021).
23. Shin, D., Yamamoto, Y., Brady, M. F., Lee, S. & Haynes, J. A. Modern data analytics approach to predict creep of high-temperature alloys. Acta Mater. 168, 321–330 (2019).
24. Verma, A. K. et al. Mapping multivariate influence of alloying elements on creep behavior for design of new martensitic steels. Metall. Mater. Trans. A 50, 3106–3120 (2019).
25. Chang, Y.-J., Jui, C.-Y., Lee, W.-J. & Yeh, A.-C. Prediction of the composition and hardness of high-entropy alloys by machine learning. JOM 71, 3433–3442 (2019).
26. Peng, J., Yamamoto, Y., Hawk, J. A., Lara-Curzio, E. & Shin, D. Coupling physics in machine learning to predict properties of high-temperatures alloys. npj Comput. Mater. 6, 141 (2020).
27. Shen, C. et al. Physical metallurgy-guided machine learning and artificial intelligence design of ultrahigh-strength stainless steel. Acta Mater. 179, 201–214 (2019).
28. Bhattacharya, S. K., Sahara, R. & Narushima, T. Predicting the parabolic rate constants of high-temperature oxidation of Ti alloys using machine learning. Oxid. Met. 94, 205–218 (2020).
29. Sedgwick, P. Pearson’s correlation coefficient. BMU 345, e4483 (2012).
30. Reshef, D. N. et al. Detecting novel associations in large data sets. Science 334, 1518–1524 (2011).
31. Freedman, D. A. Statistical Models: Theory and Practice (Cambridge Univ. Press, 2009).
32. MacKay, D. J. Bayesian interpolation. Neural Comput. 4, 415–447 (1992).
33. Tipping, M. E. Sparse Bayesian learning and the relevance vector machine. J. Mach. Learn. Res. 1, 211–244 (2001).
34. Altman, N. S. An introduction to kernel and nearest-neighbor nonparametric regression. Am. Stat. 55, 175–185 (1992).
35. Barandiaran, I. The random subspace method for constructing decision forests. IEEE Trans. Pattern Anal. Mach. Intell. 20, 832–844 (1998).
36. Awad, M. & Khanna, R. Efficient Learning Machines: Theories, Concepts, and Applications for Engineers and System Designers. (Apress, 2018).
37. Pint, B. A., Haynes, J. A. & Armstrong, B. L. Performance of advanced turbocharger alloys and coatings at 850–950 °C in air with water vapor. Surf. Coat. Technol. 219, 90–95 (2013).
38. Pillai, R., Romedene, M., Haynes, J. A. & Pint, B. A. Oxidation behavior of candidate NiCr alloys for engine exhaust valves: part I—effect of minor alloying elements. Oxid. Met. 95, 157–187 (2021).
39. Pieraggi, B. Calculations of parabolic reaction rate constants. Oxid. Met. 27, 177–185 (1987).
40. Hindam, H. & Whittle, D. Microstructure, adhesion and growth kinetics of protective scales on metals and alloys. Oxid. Met. 18, 245–284 (1982).
41. Poquillon, D. & Monceau, D. Application of a simple statistical spalling model for the analysis of high-temperature, cyclic-oxidation kinetics data. Oxid. Met. 59, 409–431 (2003).
42. Pint, B. A., Terrani, K. A., Brady, M. P., Cheng, T. & Keiser, J. R. High temperature oxidation of fuel cladding candidate materials in steam–hydrogen environments. J. Nucl. Mater. 440, 420–427 (2013).
43. Kuner, M. C., Romedene, M., Fernandez-Zelaia, P. & Dreyerpondt, S. Quantitatively accounting for the effects of surface topography on the oxidation kinetics of additive manufactured Hastelloy X processed by electron beam melting. Addit. Manuf. 36, 101431 (2020).
44. Pint, B. A., Pillai, R., Lance, M. J. & Keiser, J. R. Effect of pressure and thermal cycling on long-term oxidation in CO2 and supercritical CO2. Oxid. Met. 94, 505–526 (2020).
45. Raffaitin, A., Monceau, D., Andrieu, E. & Crabos, F. Cyclic oxidation of coated and uncoated single-crystal nickel-based superalloy MC2 analyzed by continuous thermogravimetric analysis. Acta Mater. 54, 4473–4487 (2006).
46. Gheno, T., Monceau, D. & Young, D. J. Kinetics of breakaway oxidation of Fe–Cr and Fe–Cr–Ni alloys in dry and wet carbon dioxide. Corros. Sci. 77, 246–256 (2013).
47. Monceau, D. & Poquillon, D. Continuous thermogravimetry under cyclic conditions. Oxid. Met. 61, 143–163 (2004).
48. Dryepondt, S., Turan, J., Leonard, D. & Pint, B. A. Long-term oxidation testing and lifetime modeling of cast and ODS FeCrAl alloys. Oxid. Met. 87, 215–248 (2017).
49. Zurek, J. et al. Growth and adherence of chromia based surface scales on Ni-base alloys in high-and low-pO2 gases. Mater. Sci. Eng., A 477, 259–270 (2008).
50. Pint, B. A. Addressing the role of water vapor on long-term stainless steel oxidation behavior. Oxid. Met. 95, 335–357 (2021).
51. Wallwork, G. The oxidation of alloys. Rep. Prog. Phys. 39, 401–485 (1976).
52. Giggins, C. & Pettit, F. S. Oxidation of Ni-Cr-Al alloys between 800 degrees and 1200 degrees C. Trans. Met. Soc. AIME 245, 2495–2507 (1969).
53. Giggins, C. S. & Pettit, F. S. Oxidation of Ni-Cr-Al alloys between 1000 degrees and 1200 degrees C. J. Electrochem. Soc. 118, 1782–1790 (1971).
54. Stott, F. H. Influence of alloy additions on oxidation. Mater. Sci. Technol. 5, 734–740 (1989).
55. Croll, J. E. & Wallwork, G. R. The high-temperature oxidation of iron-chromium-nickel alloys containing 0–30% chromium. Oxid. Met. 4, 121–140 (1972).
56. Rash, J. E. & Sutcliffe, J. V. River flow forecasting through conceptual models part I—a discussion of principles. J. Hydrol. 10, 282–290 (1970).
57. Breiman, L. Random forests. Mach. Learn. 45, 5–32 (2001).
58. Peng, J., Lee, S., Williams, A., Haynes, J. A. & Shin, D. Advanced data science toolkit for non-data scientists—A user guide. CALPHAD 68, 101733 (2020).
59. Geenens, G. Curse of dimensionality and related issues in nonparametric functional regression. Stat. Surv. 5, 30–43 (2011).
60. Dean, A., Voss, D. & Dragojlić, D. Design and Analysis of Experiments Vol. 1 (Springer, 1999).
61. Pint, B. A., Tortorelli, P. F. & Wright, I. G. Effect of cycle frequency on high-temperature oxidation behavior of alumina-forming alloys. Oxid. Met. 58, 73–101 (2002).
62. Pearson, K. & Lee, A. On the laws of inheritance in man: I. Inheritance of physical characters. Biometrika 2, 357–462 (1903).
63. Lee, S., Peng, J., Williams, A. & Shin, D. ASCENDS: advanced data SCiENce toolkit for non-data scientists. J. Open Source Softw. 5, 1656 (2020).
64. James, G., Witten, D., Hastie, T. & Tibshirani, R. An Introduction to Statistical Learning Vol. 112 (Springer, 2013).

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AUTHOR CONTRIBUTIONS
D.S. conceived the study. R.P., M.R., and B.A.P. curated and prepared the dataset. J.P. performed correlation analysis and machine learning training. Commercial alloys relevant to the exhaust valve environment were selected by G.M., B.A.P., and J.A.H. All authors analyzed the data. J.P. and D.S. drafted the manuscript. All authors reviewed and edited the manuscript.

COMPETING INTERESTS
The authors declare no competing interests.

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