Selective functionalization of patterned glass surfaces

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Tailored writing and specific positioning of molecules on nanostructures is a key step for creating functional materials and nano-optical devices, or interfaces for synthetic machines in various applications. We present a novel approach for the selective functionalization of patterned glass surfaces with functional probes of any nature. The presented strategy is optimized for imaging fluorophore labeled nanostructures for (single-molecule) fluorescence microscopy. The first step in the protocol is coating a glass surface, here a microscope cover slide, with a 60 nm thick diamond-like carbon film. Subsequently, the pattern is defined by either writing silicon oxide on the coating with a focused electron beam, or by etching the coating with a focused ion beam to expose the glass surface. Finally, the pattern is silanized and functionalized. We demonstrate the selective binding of organic fluorophores and imaging with high contrast, especially in total-internal-reflection mode. The presented approach is flexible and combines bottom-up assembly with high-resolution lithography on glass cover slides to precisely position and image functional molecules of any type.

1. Introduction

Smart materials can change their properties under the influence of external stimuli and can potentially be used in medical, civil and military applications. Nature is a source of inspiration for developing such smart materials, since biological systems usually contain numerous complex functional components. For instance, myosin molecules, a broad family of motor proteins, generate force for muscle contraction and guanosine nucleotide-binding proteins transmit signals across cell membranes, acting as molecular switches. The ability to mimic this allows us to produce advanced materials with responsive surfaces.

Recently, an increasing number of such functional molecules has been synthesized in laboratories, for instance, artificial molecular motors or electrical switches. Such molecules have been used for the light induced changing of the hydrophobicity, the conductance, for inducing macroscopic mechanical deformation or for attaching cells or proteins on surfaces for therapeutic applications.

However, if we want to exploit these molecules and functionalities for complex devices, we will have to interface the molecules with a surface and/or other components. One important issue is to position them with high precision. For instance, the precise positioning of functional molecules is necessary in nanophotonics to couple optically active molecules with structures like nanoantennas or mechanical resonators. But while top-down lithography has progressed from the micrometer scale down to the nanometer scale and even below, several other issues remain to be solved. Do the molecules retain their functionality at that position on the surface? How do we verify both position and functionality at the molecular level? What are the practical requirements for constructing such systems?

Several strategies are available for selective deposition of functional molecules at the submicrometer scale. Most of them rely on self-assembled monolayers (SAMs) that cover a sample surface, comparable to a carpet. The SAM is then locally modified using electron beam lithography or scanning probe microscopy. The created pattern can be decorated with functional groups, while the remaining SAM carpet prevents undesired and unselective binding on the rest of the surface. Other strategies, such as dip-pen lithography and single-molecule cut-and-paste enable the selective decoration of surfaces with functional groups without any pre-treatment of the surface. We have recently presented a purely additive functionalization technique, which we called molecular assembly controlled by electron beam induced deposition (MACE-ID). MACE-ID allows us to precisely deposit siliconoxide (SiOx) structures of defined size and shape on a substrate of choice, which can be selectively decorated with functional molecules. We demonstrated that it is possible to decorate structures as small as 25 nm selectively with fluorophores.
However, our studies have shown that it is desirable for fluorescence applications to significantly increase the signal-to-noise ratio during fluorescence imaging of MACE-ID structures. In our previous work the SiO$_x$ features were written on Si wafers and decorated with the fluorophore ATTO655. Since Si wafers are not optically transparent, they were imaged inverted on a microscope cover slide using a “sandwich” approach; the Si wafer was placed on a cover slide with the patterned surface facing down, being immersed in a buffer solution. This approach has several limitations. It is difficult to handle the sample and to change the buffer composition (necessary for changing the photophysical properties of fluorophores$^{28,32,33}$) in the thin liquid film between the wafer and the cover slide. Most importantly, however, imaging is only possible in the epi-fluorescence mode. Here the signal-to-signal-to-background and -noise values are value is much lower than that in total internal reflection imaging mode (TIRF), used for e.g., single-molecule$^{28}$ and localization-based super-resolution microscopy.$^{29–33}$

To overcome these limitations, we here extend MACE-ID and present an alternative strategy that allows selective functionalization of nanopatterned microscope glass slides. The general strategy is shown in Fig. 1. A glass cover slide is coated with a thin layer of diamond-like carbon (DLC), see Fig. 1a. A pattern is defined on the coated glass by either locally depositing SiO$_x$ with an electron beam (MACE-ID) or locally removing the DLC with an ion beam (Fig. 1b). In subsequent steps, the sample is silanized (Fig. 1c) and functionalized (Fig. 1d) with e.g., fluorophores. We demonstrate that the imaging of selectively labeled patterns with total internal reflection fluorescence (TIRF) microscopy is possible with these two strategies. Our new approach is flexible in the choice of (fluorescent) labels and gives precise control over the position of the molecular species.

2. Materials and methods

Unless stated otherwise, all chemical reagents were purchased from Sigma Aldrich and used as received without further purification. Standard microscope cover slides (Marienfeld, no. 1.5H, high precision) are used as substrates for imaging and functionalization.

2.1. Sample preparation

DLC coating procedure. The glass cover slides are cleaned before coating. The slides are cleaned by sonication in a series of liquids (1% Helmanex III solution, acetone, ethanol and finally 1 M KOH), rinsing with milliQ water after each step.$^{34}$ Subsequently, the slides are dried and cleaned in a plasma cleaner (Plasma Etch, PE-25-JW) for 10 min. When stored the cover slides are kept under vacuum in a desiccator.

The DLC film is deposited with a Teer UDP400/4 closed field unbalanced magnetron sputtering system with all four magnetrons powered off. A pulsed DC power unit (Pinnacle plus, Advanced Energy) is used for substrate biasing, operating at 250 kHz and 87.5% duty cycle. The base pressure of the vacuum chamber before the coating process is $2 \times 10^{-6}$ mbar.

Fig. 1 Illustration of the fabrication process. (a) The glass surface is coated with DLC by plasma enhanced physical vapour deposition. (b) The glass is patterned by either focused electron beam-induced deposition or ion beam induced etching. (c) Surface activation by silanization. (d) Functional groups are covalently linked to the activated structures.
Prior to the coating process, the glass slide substrates are cleaned by an Ar plasma for 15 min at a bias voltage of $-400$ V. Immediately after the plasma cleaning treatment, the DLC film is deposited by plasma activated chemical vapor deposition at a pulsed DC bias voltage of $-600$ V for 11 min to reach a film thickness of $\sim 60$ nm. The ratio of the flow rates of argon and acetylene is set at $\text{Ar} : \text{C}_2\text{H}_2 = 3 : 2$ at a constant pressure of $3 \times 10^{-3}$ mbar. The purity of argon and acetylene gasses is 5.0 and technical grade, respectively.

**Focused electron beam writing.** The local deposition of silicon oxide ($\text{SiO}_x$) is done using focused electron beam-induced deposition (see Fig. 1b). The lithography platform is a focused ion beam (FIB)/scanning electron microscope, SEM (Lyra-XM FIB/SEM dual beam microscope, TESCAN, CZ) equipped with a Schottky field emission cathode in combination with a gallium FIB column. The precursor for the deposition of $\text{SiO}_x$ is 2,4,6,8,10-pentamethylcyclopentasiloxane, a volatile liquid at room temperature. The vapour is introduced into the sample chamber close to the target through a multigas injection system. The electron source is operated between 10 keV and 30 keV. The beam currents are measured with a Faraday cup system. The electron source is operated between 10 keV and 30 keV. The beam currents vary between 1 pA for the smallest details and 10 nA for larger details and the Ga$^+$ ion beam is operated at 30 keV. The beam currents are measured with a Faraday cup (TESCAN) proprietary software without the use of a beam blanker.

**Focused ion beam etching.** A Lyra-XM FIB/SEM dual beam microscope is used for focused Ga$^+$ ion beam patterning. Ion beam currents vary between 1 pA for the smallest details and 10 nA for larger details and the Ga$^+$ ion beam is operated at 30 keV. The patterns are defined using TESCANs proprietary software. The smallest ion beam spot sizes are in the order of 50 nm (as specified by the manufacturer). The depth of the etched pattern is defined by varying the number of scans.

**Surface activation by silanization.** The selective functionalization is achieved by covalently linking an amine-containing silane, 3-aminopropyl-dimethyl-ethoxysilane, APDMES, (Acros-Organo) to either the e-beam deposited $\text{SiO}_x$ (option II, Fig. 1c) or the glass exposed by the ion beam etching (option I, Fig. 1c).

**Labeling of the silane layer.** The direct labeling of the patterned and silanized glass is shown schematically in Fig. 2a. The sample is immersed for 10 min at room temperature in a 100 nM solution of ATTO655-N-hydroxysuccinimide, ATTO655-NHS (ATTO-TEC GmbH) and 100 mM aqueous NaHCO$_3$ buffer (pH 8). For better solubility and chemical stability of the NHS-ester, 10 nmol of the dye is pre-dissolved in water free dimethylsulfoxide and diluted in the NaHCO$_3$ buffer. After the labeling the structure is rinsed with ethanol and MilliQ water.

**Surface passivation with polyethylene glycol (PEG).** Unspecific and noncovalent binding of functional molecules (e.g., fluorophores) can occur on the e-beam deposited SiO$_x$ or on the glass exposed by the focused ion beam etching. To assure that the patterned areas are decorated only with specifically (covalently) bound molecules, the patterned areas are passivated. We use polyethylene glycol (PEG) as passivation layer, which is depicted schematically in Fig. 2b (dark blue). The silanized microscope cover slide is incubated at room temperature with a solution of 1 mg mL$^{-1}$ biotin–PEG-succinimidy valerate (MW 5000, Laysan Bio, Arab, AL) and 6 mg mL$^{-1}$ methoxy-PEG-succinimidyl valerate (MW 5000, Laysan Bio, Arab, AL) dissolved in 100 mM NaHCO$_3$ buffer at pH 8. After incubation overnight any unbound PEG/PEG-biotin is removed by rinsing with MilliQ water. The pegylated sample is blow dried and stored in a desiccator.

**Surface passivation with BSA.** The patterned glass can also be prepared for labeling using the unspecific binding of bovine serum albumin (BSA) on glass (Fig. 2, dark blue). For this functionalization and passivation strategy the unsilanized

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![Fig. 2](image-url) Schematic representation of the labeling strategies, without (a) and with (b) passivation. The dimensions are not to scale.
structure is incubated at 4 °C with a 6.25 μg ml⁻¹ BSA and 1.25 μg ml⁻¹ BSA-biotin in PBS overnight. After passivation it is rinsed three times with PBS.

**Labeling of the passivation layer.** The patterned samples passivated with PEG/PEG-biotin or BSA/BSA-biotin are labeled with the following fluorophore derivatives: Cy5-streptavidin (Invitrogen) or pre-annealed dsDNA labeled with ATTO655 (Fig. 2b). Using direct streptavidin labelling, the passivated glass is exposed to 5 nM solution of Cy5-streptavidin dissolved in PBS for 2.5 min at room temperature. In the case of labeling with dsDNA, samples are incubated with 0.2 mg ml⁻¹ neutravidin (Invitrogen) for 10 min. Unbound neutravidin molecules are removed by washing the structure with PBS buffer, before incubating it with 1 nM dye-labeled DNA for 5 min. After the labeling, all structures are washed 3 times in PBS and stored in PBS.

**DNA annealing.** To specifically label the patterned areas with DNA, we use two complementary 40-mer oligonucleotides. One contains the linking biotin-unit, referred to as DNA1 (biotin-5’-CGT ATA GCT ATG CAA TAT AAG TGT AAG GAA TCG AAT ATT A-C14 at a concentration of 100 μM and pre-annealed in a heating block (Eppendorf, Mastercycler Pro). For this 100 μL of 1 μM solution of DNA1 and DNA2 is heated to 98 °C for 4 min and cooled to 4 °C at a rate of 1 °C min⁻¹ in an annealing buffer (500 mM sodium chloride, 20 mM TRIS–HCl, 1 mM EDTA, pH = 8).

**Imaging buffers.** All measurements are performed at room temperature working in an aqueous imaging buffer, based on phosphate-buffered saline (PBS) at pH 7.4. Experiments with ATTO655 are carried out in PBS only. When the fluorophore Cy5 is used, 1 mM Trolox is added and oxygen is removed from the buffer using an enzymatic oxygen scavenging system consisting of 0.1 mg ml⁻¹ glucose oxidase, 40 μg ml⁻¹ catalase, 0.2 mM tris (2-carboxyethyl)phosphine hydrochloride (TCEP), 0.1 g ml⁻¹ glucose and 10% (v/v) glycerine.

### 2.2. Sample characterization

**Ensemble absorption and fluorescence spectroscopy.** Absorption spectra in the visible and ultraviolet spectral range are recorded using a spectrophotometer (Jasco, V-640). The patterned cover slides are placed perpendicular to the beam path. Emission spectra are taken using a spectrophotometer (Jasco, FP-8300) with an angle of 90 degree between the excitation and detection path. Both excitation and detection are done under an angle of 45 degrees with respect to the sample surface.

**Atomic force microscopy.** The topology of patterned samples is inspected using an atomic force microscope (Nanoscope IIIa, Digital Instruments), operated in tapping mode. An uncoated n-doped Si cantilever (Veeco Instruments SAS) with a spring constant of 20–80 N m⁻¹ and a characteristic frequency of 300–366 kHz was used.

**Total internal reflection fluorescence imaging.** Total internal reflection fluorescence (TIRF) imaging is carried out on an inverted microscope body (Olympus IX-71) equipped with an oil immersion objective lens (UPlanSapo 100×, NA 1.4 Oil, Olympus). Additional lenses are inserted to achieve an overall magnification of 160X. Under these conditions, one pixel images an area of 100 × 100 nm² on the sample. ATTO655 and Cy5 are excited with a 637 nm solid state laser (Obis, Coherent) at an intensity of ~0.2 kW cm⁻². The laser beam is spectrally filtered using a cleanup filter (zet637/10×, Chroma) and coupled into the microscope objective via a dualband beamsplitter (2532/642rpc, Chroma). Fluorescent light passing through an emission filter (ET700/75m, Chroma) is detected using a CCD camera with an electron multiplier (ImagEM, Hamamatsu). TIRF or low-angle oblique (LAF) illumination is used to minimize the fluorescence background. For static
Fig. 4  FEB SiO$_x$ structures characterized by SEM and TIRF imaging. (a) The activated surface is labeled directly with ATTO655-NHS. (b) The SiO$_x$ is passivated with biotin/BSA and labeled with ATTO655-linked, biotinylated DNA. (c) The SiO$_x$ is passivated with biotin/PEG via neutravidin and labeled with ATTO655-linked, biotinylated DNA. The pattern is only weakly visible in TIRF. (d) The same pattern from (c) is visible in epi-fluorescence.
images, 100 frames are recorded at a frame rate of 10–30 Hz and averaged to suppress background noise.

Patterned cover slides are imaged with the patterned side facing away from the objective. A microscopic chamber gasket (LabTek, Nunc) is placed on top of the glass to enable the addition of the imaging buffer and the removal of oxygen.

3. Results and discussion

Before using DLC-coated glass for the assembly of fluorescent nanostructures, we characterized the optical properties. Fig. 3 shows the transmission and emission spectra of uncoated and DLC coated cover slides obtained with an optical spectrophotometer. As observed in Fig. 3a, the transmission is about 90% for the uncoated glass cover slide in the whole visible/near-infrared range. In comparison, a DLC-coated cover slide has a transmission of about 50% (at 637 nm, the excitation wavelength in TIRF imaging). The emission properties of the DLC coated glass are studied by exciting the sample with 637 nm light in a spectrofluorometer. As reported in the literature, the coating itself shows no fluorescence, but a detectable signal contribution due to Raman scattering by the glass slide is seen, which is further enhanced and shifted by the DLC coating (Fig. 3b).

In addition to its optical activity, the DLC coating will act as a spacer in TIRF imaging. TIRF imaging relies on the evanescent, near-field wave that is generated at the glass interface. Since the evanescent field decays exponentially from the interface (in about 100–200 nm), the DLC coating effectively reduces the excitation of fluorophores sitting on top of it. Based on these observations, we expect the imaging signal for patterns prepared with MACE-ID to be limited when combined with a DLC coating.

Fig. 4a shows the results of the functionalization using an adaption of the MACE-ID strategy, i.e. the labeling of electron beam written nanostructures on a coating (Fig. 1b, option II). SiO$_x$ squares deposited on top of the diamond-like carbon (DLC) are faintly visible in the SEM image, as the contrast between the DLC and SiO$_x$ thin film is low. The sample is silanized with 3-aminopropyl-dimethyl-ethoxyisilane (APDMS) and labeled with the fluorophore ATTO655-NHS. When imaged in TIRF mode, the contrast is high. The background intensities in the image are low, while the SiO$_x$ deposits are clearly visible.

We observe in our experiments that, when patterned glass is silanized and functionalized directly, nonspecific binding occurs. On the locally exposed surface there are not only fluorophores covalently bound to the silanes (as intended), but also fluorophores that stay on the glass surface through physisorption or chemisorption. To prevent unspecific and noncovalent binding of fluorophores to the surface, we first passivated the SiO$_x$ deposits with BSA (Fig. 4b). Fig. 4b shows a passivated sample with BSA/BSA-biotin and subsequent labeling with ATTO655-linked dsDNA oligomers, immobilized via neutravidin–biotin interactions. The SEM image shows the SiO$_x$ squares written with the electron beam, but TIRF microscopy reveals a significant background signal and a poor signal from the labeled structures. In addition, fluorescent islands are visible outside the patterned area, suggesting that aggregates of the BSA/DNA-fluorophore are present on DLC.

**Fig. 5** The etched depth as function of the number of scans. (a) SEM image showing a series of 10 by 10 μm squares with increasing amount of scans. From the contrast reversal it is concluded that the DLC coating is fully removed at 150 scans. (b) Depth profile of the corresponding, etched terraces measured using the AFM. The DLC layer (depicted in grey) is about 60 nm thick. From about 50 scans onwards DLC is removed from the glass and shows a linear sputtering rate of about $-0.8 \pm 0.02$ nm/scan.
Fig. 4c shows the results of the second approach, in which we passivate the SiO$_x$ deposits with PEG/PEG-biotin and label them with ATTO655-linked dsDNA via neutravidin–biotin interaction. The SEM image shows two groups of deposited SiO$_x$ lines written by MACE-ID with 4000 and 8000 scans each. The 50 nm wide lines are spaced at 800, 400 and 250 nm. TIRF microscopy shows no
fluorescent background around the written patterns, but also shows no detectable signal of the labeled deposits themselves. To verify that the SiO$_x$ are labeled, the written glass slide is inverted and inspected in epi-fluorescent mode. Although the signal-to-noise ratio is poor (due to the buffer layer used in epi-fluorescence mode), Fig. 4d indeed shows that the SiO$_x$ patterns are labeled via biotin–neutravidin in combination with PEG as the passivation layer.

The results show that the BSA passivation layer physisorbs to the surface and does not selectively bind to the SiO$_x$ deposits. And while the PEG/PEG-biotin is bound covalently to the activated surface and allows controlled and selective labeling, from Fig. 4c it is clear that the SiO$_x$ deposits are not observed in TIRF. From these results we conclude that it is caused by the DLC coating and the PEG/PEG-biotin; together they place the

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**Fig. 7** Specific labeling by ATTO655-labeled DNA via biotin–neutravidin interaction on a FIB patterned glass coated with PEG/PEG-biotin. (a) Bright field image. TIRF image of the same surface exposed to (b) 2 nM ATTO655-labeled DNA, (c) 0.2 mg ml$^{-1}$ neutravidin and (d) 2 nM ATTO655-labeled DNA after incubation with neutravidin. Biotinylated DNA binds exclusively after addition of neutravidin to the surface. (e) The intensity averaged over 512 $\times$ 512 pixels as a function of time, showing the increasing labeling with incubation time.
fluorophore relatively far from the glass interface and outside the evanescent field of excitation. This explains the absence of fluorescence intensity in Fig. 4c. In addition to this, the DLC coating attenuates the fluorescence signal of any labeled structure that sits on top of the coating due to its optical properties (as shown in Fig. 3). To exploit the optical properties of the DLC coating (rather than to be limited by it) we locally remove the DLC using a focused ion beam (Fig. 1b, option I). If a pattern is defined in the DLC layer, the DLC layer itself helps to reduce the background signal of unspecifically bound fluorophores. Based on the data in Fig. 3a we conclude that the coating reduces the background signal significantly compared to the signal of actively etched patterns on glass. Fig. 5 shows the etched depth of 10 by 10 μm² squares as a function of the number of scans at a beam current of 10 nA. The thickness of the DLC coating is estimated from the SEM images; when the glass surface is exposed the contrast reverses due to charging. In Fig. 5a it is observed that this happens at 150 scans. A characterization using AFM reveals the depth profile within the structure as a function of the sputtering dosis (Fig. 5b). For a small amount of scans there is some deposition, rather than etching. We speculate that this is due to a local roughening of the surface, combined with deposition due to minor contamination in the vacuum system. However, etching is observed with proceeding scans (etch rate is about −0.8 ±/− 0.02 nm/scan). This data shows that 150 scans are sufficient to fully etch through the ~60 nm DLC layer and expose the glass.

Fig. 6 shows patterns etched into the DLC coating using the focused ion beam etching of the DLC coating, imaged using the SEM and (after functionalization) TIRF microscopy. We show functionalization without (Fig. 6a) and with glass passivation (Fig. 6b–d). The SEM images (Fig. 6a) reveal regularly spaced lines and crosses. Investigating the sample with TIRF microscopy a field of excitation. This explains the absence of fluorescence intensity in Fig. 4c. In addition to this, the DLC coating attenuates the fluorescence signal of any labeled structure that sits on top of the coating due to its optical properties (as shown in Fig. 3). To exploit the optical properties of the DLC coating (rather than to be limited by it) we locally remove the DLC using a focused ion beam (Fig. 1b, option I). If a pattern is defined in the DLC layer, the DLC layer itself helps to reduce the background signal of unspecifically bound fluorophores. Based on the data in Fig. 3a we conclude that the coating reduces the background signal significantly compared to the signal of actively etched patterns on glass. Fig. 5 shows the etched depth of 10 by 10 μm² squares as a function of the number of scans at a beam current of 10 nA. The thickness of the DLC coating is estimated from the SEM images; when the glass surface is exposed the contrast reverses due to charging. In Fig. 5a it is observed that this happens at 150 scans. A characterization using AFM reveals the depth profile within the structure as a function of the sputtering dosis (Fig. 5b). For a small amount of scans there is some deposition, rather than etching. We speculate that this is due to a local roughening of the surface, combined with deposition due to minor contamination in the vacuum system. However, etching is observed with proceeding scans (etch rate is about −0.8 ±/− 0.02 nm/scan). This data shows that 150 scans are sufficient to fully etch through the ~60 nm DLC layer and expose the glass.

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4. Conclusion

We present a novel approach to combine the bottom-up assembly and high-resolution lithography to position functional molecules on glass. The glass is coated with a 60 nm thick diamond-like carbon film. A pattern is defined by either writing SiO2 on the film with a focused electron beam, or exposing the underlying glass by focused ion beam etching. The patterns are silanized and labeled with a variety of functional groups, such as (but not limited to) ATTO655 and Cy5. The approach enables imaging of the patterns with TIRF microscopy. We demonstrate that the labeling is selective and that the patterns can be imaged with high signal-to-noise ratios.

The approach enables us to both position functional groups on a surface and to verify their positions and surface concentrations accurately. Our approach promises to benefit, amongst others, single-molecule biophysics studies and studies using zero-mode waveguides. The specific positioning of single fluorescently labeled molecules, such as DNA, proteins, or viruses simplifies data analysis and provides a solution for the problem of unspecific binding in single-molecule experiments.

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