A Green Synthesis of Grpahene Based Composite for Energy Storage Application

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Abstract—In this study, graphene-molybdenum oxide composite materials were prepared via green hydrothermal synthesis method and evaluated as supercapacitor electrodes. The morphology and structure of the composite were examined by using Scanning Electron Microscopy (SEM), Raman spectroscopy. The electrochemical performances of the composite were evaluated by cyclic voltammetry (CV), galvanostatic charge-discharge (CD) method, and electrochemical impedance spectroscopy (EIS). The electrochemical results show that the composite electrodes possess improved specific capacitance of 122 F/g at a scan rate of 5 mV/s, which is about 22% higher that of pure graphene. Additionally, the composite electrodes exhibit good capacitive properties and a high specific energy with superior capacitive retention after 1000 cycles. In contrast to the previously reported systems that are usually complicated and costly, the present work potentially provides a readily scalable technological platform for economic mass production of energy storage devices.

Keywords: Supercapacitors; composite; graphene; molybdenum oxide; hydrothermal

I. INTRODUCTION

Global warming, air-pollution emission, rising oil prices and diminishing supply of fossil fuel have been increasingly driving the world towards the development of clean and environmentally friendly high power energy resources. Electrochemical capacitors, also known as supercapacitors, have attracted considerable attention over the past decades as the renewably-generated energy can be stored inside. Besides, it also exhibits higher power density, longer cycle life and moderate energy density. Innovative electrode material with desirable properties coupled with low cost and sustainability is the key for realizing high-performance supercapacitors.

Graphene, a two-dimensional nanosheet of graphite arranged in a honeycomb crystal structure possess excellent electrical conductivity, large theoretical surface area, good structural flexibility and superior chemical tolerance is commonly coupled with transition metal oxide that is high in specify capacity in designing the electrode material in the supercapacitor. These include Graphene-tungsten oxide, graphene-nickel oxide.

A number of transition metal oxides were also reported as the electrode material in the supercapacitor includes ruthenium oxide (RuO₂), nickel oxides (NiO), cobalt oxide (Co₃O₄), molybdenum oxide (MoO₃). Molybdenum oxide is a potential candidate in the electrode material in the supercapacitor because of its layered structure that capable in ion insertion/extraction, high electrochemical performance, low electrical resistivity and excellent stability.

The choice of the electrolyte is one of the most influential factors that could affect the overall electrochemical performance of the supercapacitor. Aqueous medium electrolytes including sulphuric acid and potassium hydroxide are of good candidates in providing higher power density due to the high ionic concentration. However, these aggressive electrolytes will cause severe corrosion on the energy storage devices. In this work, mild and non-flammable aqueous electrolyte (neutral sodium sulfate solution) which is beneficial to the users, the environment and the cycle life of electrodes were employed in the system.

II. MATERIALS AND METHODS

Herein, graphene-MoO₃ composites were prepared via green hydrothermal synthesis method by using ammonium molybdate and graphene oxide as the precursors. Acetic acid was used as the reducing agent in the hydrothermal process as it is relatively non-toxic in the sense that it is not harmful to the aquatic life.

Graphene Oxide (GO) and ammonium molybdate were purchased from Ugent Tech Sdn. Bhd. and MedigeneSdn. Bhd respectively. Acetic acid (99.8%) was purchased from Bendosen. Sodium sulfite used as the electrolyte was purchased from R&M Chemicals.

Briefly, 5 mg of GO and 0.6 g of ammonium molybdate was dissolved into 50 ml of 5 M acetic acid solution. The resultant mixture was then ultra-sonicated for 30 minutes to ensure a homogenous GO-dispersed solution. The homogenous dispersed solution was then transferred into the teflon-lined stainless steel autoclave and kept into the heating furnace at 180°C for 8 hours with ramping rate of 10 °C/min. The composite was then obtained by centrifuging the solution before proceeding to the calcination process with 650 °C for 2 hours under nitrogen atmosphere.
The electrode material was prepared by mixing the active material with carbon black (Ketjenblack, EC-600JD from Akzo Nobel, surface area 1400 m²/g) and Polyvinylidenefluoride (PVDF) with the weight ratio of 80:15:5. The mixture was then diluted with N-methyl-2-pyrrolidinone (NMP) before coated on 10 mm x 10 mm current collector.

The morphology and structure of the composite were examined by using Scanning Electron Microscopy (SEM), Raman spectroscopy. All the electrochemical testing was conducted with two-electrode Teflon cell in 1 M of sodium sulfite solution (Na₂SO₃). Cyclic voltammetry (CV), charge-discharge testing (CD) and electrochemical impedance spectroscopy (EIS) were conducted using MetrohmAutolab PGSTAT128N potentiostat equipped with a FRA2 frequency response analyser module and FRA software.

III. RESULTS AND DISCUSSION

Fig. 1 shows the SEM image of the graphene-molybdenum oxide composite with the optimum combination ratio of the graphene oxide and ammonium molybdate, which is 1 to 12. The MoO₃ particles were populated and distributed uniformly on the surface of reduced graphene oxide (rGO), indicating the optimal ratio which could exhibit more superior energy storage capacity. In addition, the MoO₃ sandwiched between the rGO could work as the spacer that prevents therGO from agglomeration. The mechanism of the intercalation of MoO₃ between rGO could be explained in two ways. (i) The electrostatic attraction between the oxygen functional group and the molybdenum ions. (ii) The ammonium molybdate will forms a thin layer on the surface of the GO.

Fig. 2 shows the Raman spectroscopy of the composite. There are 2 dominant peaks in the spectrum which are the D peak at ~1350 cm⁻¹ and the G peak at ~1580 cm⁻¹ respectively. G peak is responsible to the E²g phonon of the graphitic materials while the D peak is enhanced by the defects (edge defects), number of layers of the rGO. On the other hand, there are peaks appeared at around 334, a 830, 874 and 929 cm⁻¹. 334 cm⁻¹ owning to the bending modes of the terminal Mo=O bond. Meanwhile, 830, 874 and 929 cm⁻¹ are attributed by either symmetric or asymmetric stretching modes of the terminal Mo=O bond. The characteristic peaks of MoO₃ particles and rGO appears together in a spectrum, confirming the formation of the composite without destroying the structure of rGO after the hydrothermal reduction process.

The CV curve of the composite obtained in the potential window from 0 V to 1 V in 1 M of Na₂SO₃ at scan rate of 5, 20, 50 and 100 mV/s was presented in Fig. 3. The specific capacitances of the materials were calculated based on the formula in Eq. (1) where I is the applied current density, dv/dt is the rate of change of the operating potential.

\[ C_p = \frac{I}{m \times \frac{dv}{dt}} \] (1)

The specific capacitance of the composite is 121.98 F/g compared to the rGO and MoO₃ which is only 95.19 F/g and 32.90 F/g respectively. This demonstrates the synergistic effect of the composite between the rGO and the MoO₃ which could be attributed by (i) high surface area of rGOwith excellent electrical conductivity (ii) the combination of the double layer and pseudo-capacitance that increase the total capacitance of the electrodes (iii) intercalation of the MoO₃ between the rGO, improving the effectiveness of the adsorption/desorption of ions on the rGO.

Fig. 4 shows the CV curve for the composite after 300, and 1000 cycles. After 1000 cycles, the composite exhibits 83% of reversible capacity retention. Meanwhile, the percentage of retention for MoO₃ is only 62%. It is expected that the MoO₃ exhibits a lower percentage of retention as the
MoO₃ electrodes experience fading as the increasing charge-discharge cycles. This causes the pulverization and cracking of the material, which result in poorer capacity. On the other hand, the improved performance observed in this paper is attributed to the rGO-MoO₃ binary composite due to the fact that rGO could store more ions on the high surface area and serve as good electrical conductors. Moreover, the rGO can work as the spacers to limit the volume expansion of the MoO₃ during the charge-discharge process. In short, high percentage of retention indicates the good charge-discharge performance.

The calculated value of energy density and power density for different scan rate were plotted in the Ragone plot as shown in Fig. 5. The composite exhibits superior energy density at 16.27 Wh/kg at power density of 306.01 W/kg. Meanwhile, the composite also exhibits excellent power density of 1323.64 W/kg as energy density of 3.52 Wh/kg. Overall, the composite demonstrate about 5 and 4 times of enhancement of the energy density and power density respectively as compared to the rGO and MoO₃.

The IR drop of the materials was evaluated via the charge-discharge test (CD). The IR drop was calculated based on Eq. (5), where \( R_i \) is the IR drop, \( \Delta V_2 \) is the different in the voltage and \( I_{\text{discharge}} \) is the current discharge.

\[
R_i = \frac{\Delta V_2}{I_{\text{discharge}}}
\]

The IR drop values for composite, individual rGO and individual MoO₃ are 21.44 Ω, 39.67 Ω and 25.10 Ω respectively. It clearly shows that the composite exhibits the lowest internal resistance due to the synergetic effect of the rGO and the MoO₃.

Electrochemical impedance spectroscopy (EIS) was conducted in order to analyze the resistance of the materials. Based on Fig. 6, it can be clearly observed that the diameter of the semicircle for rGO is the largest among all the materials, thus suggesting that rGO anode have highest contact and the charge-transfer resistance caused by the double layer capacitance is also the greatest, followed by MoO₃ and finally the composite. As a result, the composite demonstrates the lowest charge-transfer resistance. This is attributed by the flowing factors where the present of the rGO in the composite provides good conductivity. Besides, the mesh structure of the rGOs able to hold MoO₃ particles in its micropores, preventing the MoO₃ from agglomeration. As a result, it enhances the contact area between electrolyte ions and electrodes, which eventually decreases the charge-transfer resistance.

An equivalent circuit as shown in Fig. 7 was constructed in order to fit the Nyquist plot. After the simulation and fitting, it is found that the composite exhibits the lowest resistance among the electrode materials. Table 1 shows the resistance of the materials obtained from the equivalent circuit which is similar to the trend obtained in the IR drop from the CD test.
Table 1: The resistances of different electrode materials measured in ohm (Ω).

| Composite | rGO  | MoO3 |
|-----------|------|------|
| Polarization | 2.00 | 13.50 | 3.84 |
| Solution   | 0.79 | 1.81 | 0.50 |
| Total      | 2.79 | 15.31 | 4.34 |

Fig. 6: Nyquist plot of the materials

Fig. 7: Equivalent circuit for the composite to fit the Nyquist plot.

Overall, the results reveal that graphene-molybdenum oxide composite in a Na₂SO₄ neutral electrolyte exhibits excellent electrochemical performance and high cycling stability, thus displays great potential for supercapacitor applications.

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