Measuring the degree of “nanolitization” of volcanic glasses: Understanding syn-eruptive processes recorded in melt inclusions

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1. Introduction

Magma transport plays a fundamental role in the evolution of planets (van Keken, 2003) and the activity and eruption style of volcanoes (Gonnermann and Manga, 2007). Its eruption on the Earth’s surface can pose a significant threat to economic activities and human lives (Newhall et al., 2018). Silicate melt is, in most cases, the fundamental component of magma and its viscosity plays a key part in controlling magma mobilization and transport (Hammer et al., 1999; Petford, 2003; Pistolesi et al., 2011; Zhang, 1999). The viscosity of a natural silicate melts can vary by several order of magnitude as function of chemical composition (X), temperature (T), and oxygen fugacity (fO2) (Bouhifd et al., 2004; Hui and Zhang, 2007; Kolzenburg et al., 2018; Stabile et al., 2016; Webb et al., 2014). This dramatically affects the magmas transport dynamics and thus its eruptive behaviour. Changes in melt viscosity are the result of modifications in the structural interconnectivity of the melt and changes in the size of tetrahedra rings that are mostly driven by changes in X-T-fO2 (Kubik and Toplis, 2002; McMillan, 1984; Mysen, 1999; Sharma et al., 1981; Toplis et al., 1997). Therefore, understanding the X-T-fO2 dependent structure-properties relationship represents a core challenge in advancing our knowledge of magmatic and volcanic processes and can inform conceptual and computational approaches to hazard forecasting.
The structure of silicate melts consists of a three-dimensionally interconnected network of different units (Mysen and Richet, 2005 and reference therein). Tetrahedral units, the so-called Q0 species, are distinguished by the number (n) of bridging oxygens (BO), which ranges between 0 and 4, that connect Q0 species. These species are coordinated by network-former cations which, in magmatic melts, are dominantly Si4+, Al3+, and Fe3+. Iron is the most abundant 3d-transition element in volcanic systems and can be subjected to changes throughout magmatic and volcanic processes. The redox state of iron (ferrous Fe2+ and ferric Fe3+) greatly affects its role in the melt structure and thus its properties as well as the stability of Fe-bearing phases (Bouhifd et al., 2004; Dingwell et al., 1988; Giordano et al., 2015; Liebske et al., 2003; Mysen et al., 1985; Wilke, 2005). The Fe3+/Fe tot. ratio changes significantly with X-T-fO2 conditions. Several studies have shown the effect of changing the redox state of iron on the physical properties of volcanic melts (Botcharnikov et al., 2005; Bouhifd et al., 2004; Di Genova et al., 2017c; Dingwell and Virgo, 1987; Kolzenburg et al., 2018; Liebske et al., 2003; Poe et al., 2012; Stabile et al., 2016; Toplis and Carroll, 1995).

Further, a recent study by Di Genova et al. (2017a), documented how the presence of magnetite nano-crystals, termed nanolites, can alter the structure and viscosity of volcanic melts. It was demonstrated that the formation of nanolites results in a viscosity increase of up to two orders of magnitude. It is also argued that iron oxides may provide nucleation sites for bubbles (Gardner and Denis, 2004; Plese et al., 2018; Shea, 2017). Over the last few years, nanolites were also documented in products of explosive eruptions at basaltic and andesitic volcanoes (Japan, Muñin et al., 2017; Muñin and Nakamura, 2014) and the Menan Volcanic Complex (USA, Xu et al., 2017). The above summary of observations suggests that the precipitation of nanolites during magma ascent and decompression can play a crucial role in controlling the eruptive style.

Furthermore, the nano-heterogeneity of volcanic melts raises a fundamental question about our ability to model the viscosity of the melt fraction constituting the magma. Indeed, the parameterization of viscosity (e.g. Giordano et al., 2008; Hui and Zhang, 2007) often relies on melt chemistry measurements at a larger scale (i.e. microscale) than the nanocrystals and, thus, returning a seemingly homogenous bulk analysis of a heterogeneous material.

Raman spectroscopy has largely been used to infer the network connectivity of synthetic and volcanic melts and glasses (Ardia et al., 2014; Di Genova et al., 2017c; McMillan and Piirio, 1982; Mysen et al., 1982), estimate volatile content (Behrens et al., 2006; Di Genova et al., 2017b; Mercier et al., 2010; Morizet et al., 2013; Schiavi et al., 2018; Thomas, 2000), and approximate the Fe3+/Fe tot. ratio and chemistry of volcanic glasses (Di Genova et al., 2016a, 2016b; Di Muro et al., 2009). However, while recent studies focused on the effect of nanolites on viscosity and estimation of water content, so far, no study has been presented to systematically characterize the degree of nano-heterogeneity of volcanic glasses.

Here, we expand the use of Raman spectroscopy to systematically study heterogeneities in volcanic products by measuring the degree of “nanolitization” of glasses over a large range of chemical composition. To this aim, we first carefully study the effect of the most common excitation sources for Raman spectroscopy on the spectral feature related to the presence of magnetite nanolites. We then introduce a new index termed #N (i.e. nanolite number) to constrain the degree of “nanolitization” calculated from Raman spectra of glasses. This new tool enables the user to infer the effect of nanoparticles on experimental results (Di Genova et al., 2017b; Di Muro et al., 2006a; Schiavi et al., 2018) and eruptive dynamics (Di Genova et al., 2017a; Muñin et al., 2017; Muñin and Nakamura, 2014). It further enables comparative studies of results obtained from different laboratories.

2. Methodology

2.1. Starting materials

Four iron-rich rhyolites (Fsp series, from Di Genova et al., 2016b) were used to investigate the effect of laser wavelength on Raman spectra of homogenous and nanolite-free glasses characterized by different iron oxidation state (Fe3+/Fe tot. ratio ranging from 0.24 to 0.83). The effect of nanolites on Raman spectra was studied using six magnetite-nanolite-bearing glasses of different composition. Specifically, we selected three glasses from the calc-alkaline magma series: basaltic (KR3) and dacitic (HO2) glasses from Di Genova et al. (2017b) and a rhyolite (L, from Di Genova et al., 2017a). We further analysed three alkali-rich samples, namely a trachy-basalt (ETN3.8), trachyte (AM54) and latite (FR3.8) from Di Genova et al. (2014a, 2014b). The Raman spectra acquired from these samples suggested that the mineral phase is magnetite (Di Genova et al., 2017b). Magnetic-hysteresis performed on the rhyolitic sample (L, Di Genova et al., 2017a) showed that the nanolite diameter ranges between ~5 nm to ~30 nm. We acknowledge that, according to Muñin et al. (2017), particles below ~30 nm can be classified as “ultrananolite” and that the term “nanolite” is suggested for describing those from 30 nm to 1 μm in width. In this study, we use the term “nanolite” for all samples because no information on the diameter of particles dispersed in our samples is available, except for sample L. Details of the experimental procedures to synthesise these samples, together with the techniques to estimate the iron oxidation state and water content, are reported in the respective studies. The samples used for application of the developed method to natural samples were collected from the basalt vitrophyre of the Green Tuff ignimbrite (45–50 ka, Civetta et al., 1984) and Cuddia del Gallo cone (~6 ka, Mahood and Hildreth, 1986; Giocca and Landi, 2010) on Pantelleria island (Italy), respectively.

The chemical compositions of all synthetic samples are reported in Table 1, while the chemistry of natural samples is reported in Table 2. All synthetic compositions are plotted in a TAS (total alkali versus silica) diagram in Fig. 1.

2.2. Raman spectroscopy

Raman spectra were acquired using a Thermo Scientific™ DXR™xi Raman Imaging Microscope at the University of Bristol, School of Earth Sciences. The instrument is equipped with three different lasers i) blue, 455 nm diode, ii) green, 532 nm doubled Nd:YVO4 DPSS and iii) red, 633 nm HeNe. Each laser is coupled with a single grating; the spectral resolution is ~2 cm−1. Spectra collected with the blue laser (455 nm) were acquired using a 1200 lines·mm−1 grating. The green laser (532 nm) is coupled with a 900 lines·mm−1 grating, and the red laser (633 nm) is coupled with a 600 lines·mm−1 grating. It must be noted that only the green laser configuration allows for measurements from 50 to 4000 cm−1.

Spectra of synthetic samples were acquired between 50 cm−1 and 1500 cm−1 using a high confocality setting with a 100× objective, 25 μm confocal pinhole, and a laser power between 3 and 6 mW measured on the sample surface in order to avoid oxidation of the sample (Di Genova et al., 2017b). All the spectra were acquired at ~5 μm of depth where the Raman signal was found to be at its optimum. In order to maximise the signal/noise ratio, the acquisition time and repetitions ranged between 8 and 15 s and 30–50 times, respectively. Different acquisition times were chosen to compensate for the differences in photon flux per pixel due to differences in the dispersion of employed gratings. Raman spectra of natural samples were collected between 100 cm−1 and 4000 cm−1 using the green laser, with an acquisition time of 15 s and 10 repetitions. Raman maps of the natural samples were produced using a linear magnetic stage coupled with optical encoders which allows maintaining the x-y-z position repeatability to within 10 nm. The DXR™xi Raman Imaging system is equipped with...
an electron multiplied charge coupled device (EMCCD) used to amplify the signal without amplifying readout noise. This results in an extremely fast acquisition time (< 1 s) per 50 to 4000 cm\(^{-1}\) spectrum thus permitting to acquire thousands of spectra in a few minutes. This enables us to map relatively large areas (mm\(^2\)) in a satisfactory timeframe and, importantly, without heating and/or otherwise altering the sample.

All spectra were corrected for temperature and excitation line effects according to Long (1977). In order to subtract the spectra background, cubic splines were fitted through intervals devoid of peaks in the silicate region (from 1000 to 200 and 1250 to 1500 cm\(^{-1}\)) and in the water region (2750–3100 and 3750–3900 cm\(^{-1}\)). The resulting spectra were normalised to the total area. All spectra were smoothed using Fourier transforms. The water content dissolved in melt inclusions was estimated according to Long (1977). In order to subtract the spectra background, cubic splines were employed to acquire the spectra (i.e. independent of laser wavelength).

### 3. Results

Raman spectra of synthetic glasses characterized by different Fe\(^{3+}/\text{Fe}_{\text{tot.}}\) ratio (Fsp series, pantellerite) acquired using the blue, green, and red lasers are shown in Fig. 2a–d. Spectra from nanolite-bearing glasses, but they possess distinct additional vibrational contributions located between 670 and 690 cm\(^{-1}\) and ~300 cm\(^{-1}\) that are not related to the glass structure. For pure volcanic glasses, depending on the SiO\(_2\) content, the maximum Raman intensity is expected either at ~500 or ~1000 cm\(^{-1}\), depending on the dominant structural contributor (Di Genova et al., 2015; Di Genova et al., 2016b; Di Muro et al., 2009). The ~970 cm\(^{-1}\) contribution was assigned to an antisymmetric coupled mode of Fe\(^{3+}\) in four-fold coordination state, while the ~1040 cm\(^{-1}\) is characteristic of Fe\(^{2+}\)-rich pantellerite glasses (Di Genova et al., 2015; Di Genova et al., 2016a; Di Muro et al., 2009).

For comparison, we also report Raman spectra of the most oxidized of the investigated iron-rich rhyolites (Fsp1 sample, Fe\(^{3+}/\text{Fe}_{\text{tot.}}\) = 0.83) collected using the three lasers in Fig. 2a. Overall, the signal/background ratio of the spectra changes with the excitation source. As showed in Fig. 2a, the spectra acquired with the red laser exhibit the lowest signal/background ratio, while spectra collected with the blue laser show the highest signal/background ratio.

After background subtraction and area normalization (Fig. 2b–d), excluding the sample with the lowest Fe\(_2\)O\(_3\) content (Fe\(^{3+}/\text{Fe}_{\text{tot.}}\) = 0.24), the Raman spectra are dominated by the feature at ~970 cm\(^{-1}\). The spectrum of the Fe\(^{3+}\)-poor sample (Fe\(^{3+}/\text{Fe}_{\text{tot.}}\) = 0.24) shows a peak at ~1040 cm\(^{-1}\). These spectral signatures have been observed and studied extensively in spectra of simple and multicomponent systems collected with green lasers (Di Muro et al., 2009; Fox et al., 1982; Sharma et al., 1997; Virgo et al., 1983; Wang et al., 1993; Wang et al., 1995). The ~970 cm\(^{-1}\) contribution was assigned to an antisymmetric coupled mode of Fe\(^{3+}\) in four-fold coordination state, while the ~1040 cm\(^{-1}\) is characteristic of Fe\(^{2+}\)-rich pantellerite glasses (Di Genova et al., 2015; Di Genova et al., 2016a; Di Muro et al., 2009). The data presented here demonstrate that these observations hold true regardless of the excitation source employed to acquire the spectra (i.e. independent of laser wavelength).

Spectra collected from synthetic nanolite-bearing glasses are shown in Fig. 3a–f. The spectra show a broad resemblance to those of pure silicate glasses, but they possess distinct additional vibrational contributions located between 670 and 690 cm\(^{-1}\) and ~300 cm\(^{-1}\). These spectral signatures have been observed and studied extensively in spectra of simple and multicomponent systems collected with green lasers (Di Muro et al., 2009; Fox et al., 1982; Sharma et al., 1997; Virgo et al., 1983; Wang et al., 1993; Wang et al., 1995). The ~970 cm\(^{-1}\) contribution was assigned to an antisymmetric coupled mode of Fe\(^{3+}\) in four-fold coordination state, while the ~1040 cm\(^{-1}\) is characteristic of Fe\(^{2+}\)-rich pantellerite glasses (Di Genova et al., 2015; Di Genova et al., 2016b; Di Muro et al., 2009; Mercier et al., 2009; Schiavi et al., 2018). Here, the main peak position occurs either at ~670 cm\(^{-1}\) for SiO\(_2\)-rich sample (rhyolite) or at ~690 cm\(^{-1}\) for SiO\(_2\)-poor samples (basalt). Based on data from the RRUFF database (Lafuente et al., 2015) and previous studies (Di Genova et al., 2017b; Di Muro et al., 2006a), this peak can be assigned to the presence of magnetite nanolites. This spectral feature is present in all spectra independent of the laser wavelength. However, we note that the intensity of this peak changes systematically as a function of the excitation source.

### Table 2

| Sample | SiO\(_2\) | TiO\(_2\) | Al\(_2\)O\(_3\) | Fe\(_{\text{tot.}}\) | MnO | MgO | CaO | Na\(_2\)O | K\(_2\)O | H\(_2\)O | Fe\(^{3+}/\text{Fe}_{\text{tot.}}\) | N#\(_{\text{green}}\) | N#\(_{\text{red}}\) |
|--------|---------|---------|-------------|-----------------|-----|-----|-----|---------|-------|-------|-----------------|----------|----------|
| GT1    | 68.31   | 0.39    | 7.84        | 8.40            | 0.27| 0.10| 0.26| 5.71    | 4.86  | 0.03  | 0.74            | 0.00     | 0.74     |
| GT2    | 68.78   | 0.43    | 7.72        | 8.74            | 0.41| 0.08| 0.31| 5.77    | 5.07  | 0.06  | 0.90            | 0.02     | 0.90     |
| GT6    | 67.48   | 0.47    | 7.66        | 7.93            | 0.31| 0.11| 0.34| 6.05    | 4.65  | 0.09  | 1.12            | 0.04     | 1.12     |
| GT8    | 67.25   | 0.44    | 7.39        | 7.39            | 0.31| 0.08| 0.26| 6.07    | 4.97  | 0.13  | 0.01            | 0.01     | 0.01     |
| GT9    | 68.85   | 0.45    | 7.84        | 8.06            | 0.32| 0.12| 0.33| 4.73    | 4.75  | 0.07  | 1.02            | 0.04     | 1.02     |
| CDG1   | 66.34   | 0.57    | 9.50        | 8.59            | 0.32| 0.23| 1.30| 6.90    | 4.25  | 0.05  | 0.77            | 0.07     | 0.77     |
| CDG4   | 67.49   | 0.60    | 9.75        | 8.03            | 0.26| 0.14| 0.59| 6.65    | 4.60  | 0.04  | 0.61            | 0.04     | 0.61     |
source, but independently of the sample composition. Spectra acquired with the red laser (633 nm) show the highest intensity of this peak, while spectra collected using the blue laser (455 nm) display the lowest intensity.

4. Measuring the degree of “nanotilization” of synthetic glasses

As demonstrated above, the Fe$^{3+}$/Fe$_{tot}$ ratio in the glass directly correlates with the intensity of the band at ~970 cm$^{-1}$ regardless of the excitation source, while the occurrence of magnetite nanolites results in a peak at 670–690 cm$^{-1}$ whose intensity depends on the laser wavelength. Recently, Di Genova et al. (2017b) showed that the development of the nanolite peak is associated with a proportional decrease of the band at ~970 cm$^{-1}$. This implies that magnetite nanolite precipitation occurs via transfer of Fe$_2$O$_3$ from the melt to the crystal structure. Based on these observations, we introduce a new index ($N$#) to quantify this transfer and, thus, the degree of “nanotilization” of volcanic glasses (i.e. how much of the total Fe$_2$O$_3$ has precipitated into crystals from the melt). It should be noted that, according to recent TEM observations on volcanic glasses (Mujin et al., 2017), the melt depletion of the Fe$_2$O$_3$ resulting from the nucleation and growth of nanolites is mainly controlled by the crystallization of magnetite and/or pyroxene. In our synthetic samples, however, we observed the occurrence of only magnetite and thus, here, $N$# refers to the “nanotilization” controlled by the crystallization of magnetite.

For each spectrum, we calculate (Table 1) the nanolite number $N$# as follows:

$$N# = \frac{I_{670-690}}{I_{970}}$$

where $I_{670-690}$ and $I_{970}$ represent the spectral intensity of the nanolite peak (between 670 and 690 cm$^{-1}$) and the Fe$^{3+}$ band (~970 cm$^{-1}$), respectively. This index increases with increasing degree of nanotilization and is equal to zero for a nanolite-free glass. In Fig. 4a–f we report the Long- and baseline-corrected spectra of nanolite-bearing glasses collected with the three different lasers.

Fig. 5 shows the calculated $N$# as a function of the laser wavelength. Due to the variable intensity for the same sample of the nanolite peak with laser wavelength (Fig. 3a–f), $N$# changes depending on the excitation source as a result of the differing laser coupling and, thus, the signal to background ratios for the ~670–690 cm$^{-1}$ and the 970 cm$^{-1}$ band. Therefore, comparative studies need to be performed using the same laser wavelength. The basaltic sample (KR3) exhibits the highest degree of nanotilization (Table 1 and Fig. 5), followed by the latitic (FR3.8), dacitic (HO2) and rhyolitic (L) samples. On the other hand, the trachytic (AMS4) and trachytic (ETN3.8) are characterized by the lowest $N$# (Table 1 and Fig. 5).

5. Discussion: Nano-heterogeneity of natural products

Although to date little is known about the systematics of nanolite precipitation in magma, recent studies based on textural, geochemical, and experimental analyses of glasses and melts suggest that nanolites precipitate predominantly during magma ascent (Di Genova et al., 2017a; Mujin et al., 2017; Mujin and Nakamura, 2014). In order to explore the possibility of retrieving information on the changes of the melt structure during the eruption, we acquired Raman spectra of closed, partially degassed, and leaked melt inclusions (MIs). These different glasses can provide information on the structure of the melt frozen in when it crossed the glass transition temperature during the eruptive event.

Pantellerites are one of the most enigmatic peralkaline rhyolites since they are enriched in iron and characterized by an extraordinarily low viscosity, but nonetheless often feed high intensity (VEI) explosive eruptions (Civetta et al., 1984; Webster et al., 1993). In spite of the wealth of studies (Campagnola et al., 2016; Di Genova et al., 2013; Hughes et al., 2017; Lanzo et al., 2013), key issues on the fragmentation
dynamics of these magmas remain elusive. We collected pantelleritic samples erupted during two explosive eruptions of different intensity at Pantelleria island (Italy): 1) the basal vitrophyre of the caldera-

Fig. 2. (a) Long-corrected Raman spectra of the Fe$^{3+}$-rich rhyolitic glass (Fsp1 sample, Fe$^{3+}$/Fe$_{tot}$ = 0.83) collected with the three different lasers before background subtraction. The signal/background ratio increases from the red (633 nm), to green (532 nm), to blue (455 nm) laser. (b, c, d) Long- and background corrected and area normalised Raman spectra of iron-rich rhyolitic glasses (Fsp series, Table 1) characterized by varying Fe$^{3+}$/Fe$_{tot}$ ratios (numbers in the legend). Spectra were acquired using three different lasers: b) blue, c) green, d) red. With increasing the iron oxidation state of the glass, the band at ~970 cm$^{-1}$ increases independent of the excitation source. The 1040 cm$^{-1}$ contribution is dominant in Fe$_2$O$_3$-poor pantelleritic glass (see the text for more details). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

Fig. 3. Long-corrected spectra of nanolite-bearing glasses with different chemical composition acquired with the 455 nm (blue), 532 nm (green), and 633 nm (red) lasers before baseline subtraction. a) trachybasalt (ETN3.8), b) basalt (KR3), c) latite (FR3.8), d) dacite (HO2), e) trachyte (AMS4), f) calcalkaline rhyolite (L). The sample composition is reported in Table 1. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

Fig. 4. Long- and baseline-corrected and area normalised Raman spectra of the nanolite-bearing samples acquired with the 455 nm (blue), 532 nm (green), and 633 nm (red) lasers. a) trachybasalt (ETN3.8), b) basalt (KR3), c) latite (FR3.8), d) dacite (HO2), e) trachyte (AMS4), f) calcalkaline rhyolite (L). Sample compositions are reported in Table 1. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)
of the diopside (Fig. 6b), due to the presence of Fe$_2$O$_3$ in its crystalline structure and GT2 spectra (Fig. 6a). The location of Raman spectra is located at $-970$ cm$^{-1}$. The Fe$^{3+}$/Fe$_{tot}$ ratio of the Raman spectra is $-0.5$ and was constrained in reference to spectra from samples of the same composition but different iron oxidation state (Di Genova et al., 2016a; Di Muro et al., 2009).

The H$_2$O content of the glass decreases from $-1.3$ wt% for the GT1 and GT2 spectra (Fig. 6a–b) to essentially water free at the outside of the leaked melt inclusion (spectra GT4 and GT5). Furthermore, the first three spectra (GT1, GT2 and GT3) exhibit a contribution at $-670$ cm$^{-1}$ (Fig. 6b) indicating the presence of Fe-bearing nanocrystals. In the presented case the spectral signature of the host mineral (diopside) exhibits a similar contribution at $-670$ cm$^{-1}$ (Fig. 6b) that is to be evaluated in order to exclude eventual misidentification of the Raman feature. The host mineral also shows a prominent peak at $-1015$ cm$^{-1}$ and two weaker contributions at $-320$ and $-390$ cm$^{-1}$. Although the latter peaks were not observed in the glass spectra GT1, GT2 and GT3, we observed the development of a slight shoulder at $-1000$ cm$^{-1}$, a lower position with respect to the main peak of the diopside. Additionally, while the full width at half maximum (FWHM) of the Fe$_2$O$_3$ peak of the diopside (Fig. 6b), due to the presence of Fe$_2$O$_3$ in its crystalline structure, is narrow ($-15-20$ cm$^{-1}$, Lafuente et al., 2015), the FWHM of the 670 cm$^{-1}$ peak of GT1, GT2 and GT3 spectra is significantly broader ($-40-45$ cm$^{-1}$) and similar to that observed in the Fe$_2$O$_3$ nanolite-bearing glasses in Fig. 4a–f and reported by previous studies (Figs. 3 in Di Muro et al., 2006a, 2006b). This suggests the incipient crystallization of diopside, resulting in the development of hybrid features between the quenched melt structure and the crystalline structure.

Overall, the evolution of the spectral signature from GT1 to GT5 is paralleled by a decrease in the H$_2$O content of the melt which results in an increase in melt viscosity and, thus, a decrease in element diffusivity from inside the inclusion to the outside. Assuming a similar thermal path for the melt in the inclusion and the outside, this allows for a higher potential of nanolite formation in the lower viscosity, water-rich and higher diffusivity inside of the melt inclusion with respect to the outside, where the melt is efficiently quenched to a glass during the eruption. The relatively high-water content may promote oxidising conditions of the system (Humphreys et al., 2015) and, combined to fast quench rate (Di Genova et al., 2017b; Di Muro et al., 2006a), induced the formation of magnetite nanolites during degassing. This process, however, is not restricted to magnetite only and can also occur during the crystallization of other phases such as for example pyroxene and plagioclase (Mujin et al., 2017). In fact, degassing-induced crystallization of both these phases is well documented for basaltic lavas (Applegarth et al., 2013; Lipman et al., 1985) but has not been described for pantelleritic melts. This is likely due to the small scale of the crystals forming during degassing, which required new techniques for their identification, such as presented here. The spectra collected at the outside of the melt inclusion (spectra GT4 and GT5, Fig. 6b) show two peaks at $-470$ and $-510$ cm$^{-1}$ that are characteristics of Raman features in feldspar. However, these peaks are broader and shifted to a lower wavenumber ($-7$ cm$^{-1}$, Fig. 7b) compared to the spectrum of the adjacent feldspar (dashed line in Fig. 7b). This suggests incipient feldspar nucleation within the melt during ascent and/or quenching. Such pre-crystalline precipitates have been identified in experimental studies on supersaturated hydrothermal fluids crystallizing silicates at high temperature and pressure (Manning, 2018).

The formation of the diopside nanolites (Fig. 6b) due to the presence of Fe$_2$O$_3$ in its crystalline structure and GT2 spectra (Fig. 6a) is associated with a prominent peak at $-1015$ cm$^{-1}$ and two weaker contributions at $-320$ and $-390$ cm$^{-1}$. Although the latter peaks were not observed in the glass spectra GT1, GT2 and GT3, we observed the development of a slight shoulder at $-1000$ cm$^{-1}$, a lower position with respect to the main peak of the diopside. Additionally, while the full width at half maximum (FWHM) of the 670 cm$^{-1}$ peak of the diopside (Fig. 6b), due to the presence of Fe$_2$O$_3$ in its crystalline structure, is narrow ($-15-20$ cm$^{-1}$, Lafuente et al., 2015), the FWHM of the 670 cm$^{-1}$ peak of GT1, GT2 and GT3 spectra is significantly broader ($-40-45$ cm$^{-1}$) and similar to that observed in the Fe$_2$O$_3$ nanolite-bearing glasses in Fig. 4a–f and reported by previous studies (Figs. 3 in Di Muro et al., 2006a, 2006b). This suggests the incipient crystallization of diopside, resulting in the development of hybrid features between the quenched melt structure and the crystalline structure.

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The formation of the diopside nanolites (Fig. 6b) due to the presence of Fe$_2$O$_3$ in its crystalline structure and GT2 spectra (Fig. 6a) is associated with a prominent peak at $-1015$ cm$^{-1}$ and two weaker contributions at $-320$ and $-390$ cm$^{-1}$. Although the latter peaks were not observed in the glass spectra GT1, GT2 and GT3, we observed the development of a slight shoulder at $-1000$ cm$^{-1}$, a lower position with respect to the main peak of the diopside. Additionally, while the full width at half maximum (FWHM) of the 670 cm$^{-1}$ peak of the diopside (Fig. 6b), due to the presence of Fe$_2$O$_3$ in its crystalline structure, is narrow ($-15-20$ cm$^{-1}$, Lafuente et al., 2015), the FWHM of the 670 cm$^{-1}$ peak of GT1, GT2 and GT3 spectra is significantly broader ($-40-45$ cm$^{-1}$) and similar to that observed in the Fe$_2$O$_3$ nanolite-bearing glasses in Fig. 4a–f and reported by previous studies (Figs. 3 in Di Muro et al., 2006a, 2006b). This suggests the incipient crystallization of diopside, resulting in the development of hybrid features between the quenched melt structure and the crystalline structure.

Overall, the evolution of the spectral signature from GT1 to GT5 is paralleled by a decrease in the H$_2$O content of the melt which results in an increase in melt viscosity and, thus, a decrease in element diffusivity from inside the inclusion to the outside. Assuming a similar thermal path for the melt in the inclusion and the outside, this allows for a higher potential of nanolite formation in the lower viscosity, water-rich and higher diffusivity inside of the melt inclusion with respect to the outside, where the melt is efficiently quenched to a glass during the eruption. The relatively high-water content may promote oxidising conditions of the system (Humphreys et al., 2015) and, combined to fast quench rate (Di Genova et al., 2017b; Di Muro et al., 2006a), induced
(GT6, Fig. 7b) exhibits the typical features of pantelleritic glasses (Fig. 2a–d), while the spectrum collected close to the bubble surface (GT7, Fig. 7b) is characterized by the ~670 cm$^{-1}$ nanolite peak which is not present in the spectrum of the host mineral (Fig. 7b). Moreover, as reported before, the difference in Raman shift of the two peaks located at ~470, ~510 cm$^{-1}$ with respect to the host mineral spectrum, together with the absence of crystalline peaks at ~170 and 290 cm$^{-1}$ in the MIs spectra, suggest that the melt structure has undergone incipient feldspar crystallization below available resolution. We also inspected the groundmass glass of the basal vitrophyre in order to study its structure. In Fig. 8a–b we report a BSE-SEM image of 1800 $\mu$m$^{2}$ over which we acquired 1800 Raman spectra. We found that, albeit the BSE-SEM image suggests this area to be composed of homogenous glass, the Raman spectra collected from the matrix groundmass were highly heterogeneous at the nanoscale. In order to quantify and visualize the degree of nanotilization, we calculated the $N\#$ (Eq. 1) for each spectrum. Fig. 8c shows a heat map visualizing the degree of nanotilization using a colour gradient from cold (blue, nanolite-free glass, low $N\#$) to hot (red, nanolite-bearing glass, high $N\#$). The distribution of colours shows the coexistence of two domains 1) a pure, nanolite free glass and 2) an extremely nano-heterogenous glass.

In line with in situ observations of bubble nucleation in silicate melts (Masotta et al., 2014; Pleše et al., 2018 and reference therein), textural observations of volcanic products (Applegarth et al., 2013; Lipman et al., 1985) and experimental results (see Shea, 2017 for a review), the development of incipient crystalline features in the Raman spectra on the inside of vesiculating melt inclusions (Figs. 6 and 7) suggest that the nanotilization stimulated/facilitated the exsolution of volatile components of the magma by providing nucleation sites that promote bubble formation, increased the melt viscosity (Di Genova et al., 2017a) and, thereby, contribute to enhancing the explosivity of volcanic eruptions.

We now expand this analysis to the lower intensity Strombolian type eruptions from the Cuddia del Gallo eruption by including flattened pantelleritic spatters. This allows us to evaluate whether secondary processes (e.g. annealing and potential oxidation) may have induced the observed structural heterogeneities of the glass.

In Fig. 9a we show a SEM image of Mls from spatters erupted during this event. The location of the acquired Raman spectra is reported in Fig. 9a, while the spectra are reported in Fig. 9b. We analysed three
characteristic melt inclusions; 1) a closed MI (CDG1), 2) a partially open MI coexisting with a bubble (CDG2), and 3) a glass embayment containing bubbles (CDG3–4).

The Raman spectra in the region at ~3550 cm⁻¹ (Fig. A.1.) revealed that only the closed and bubble-free MI (CDG1) contains water, while the embayments (CDG2–4) are anhydrous. Overall, comparison of the results obtained from Cuddia del Gallo spatter with those from the Green Tuff vitrophyre, suggests that all the MIs from the spatters of Cuddia del Gallo (CDG1–4, Fig. 9b) are more reduced (Di Genova et al., 2016a; Di Muro et al., 2009) with respect to the MIs of the Plinian eruption. This is inferred from the extremely weak contribution at ~970 cm⁻¹ of the spatter spectra (Fig. 9b) with respect to those of the Green Tuff vitrophyre (Fig. 7a, see Di Genova et al., 2016a; Di Muro et al., 2009 for further detail on how to estimate the Fe³⁺/Fetot. ratio of glasses). Furthermore, the nanolitization of the melt was more intense during the Plinian eruption (Green Tuff, Fig. 7b and 8c) than during the low-energetic Strombolian event at Cuddia del Gallo (Fig. 9b).

Previous studies estimated the pre-eruptive magmatic water content of pantelleritic eruptions at ~4–5 wt% (Di Carlo et al., 2010; Gioncada and Landi, 2010; Kovalenko et al., 1988; Lanzo et al., 2013; Lowenstern and Mahood, 1991; Métrich et al., 2006; Neave et al., 2012; White et al., 2009). These data were collected on samples from a series of different eruptions on Pantelleria Island, including those studied here. Despite there being slight differences in pre-eruptive water content and oxygen fugacity among effusive and explosive eruptions, viscosity measurements of the effect of H₂O (Di Genova et al., 2013) and Fe³⁺/Fetot. ratio (Stabile et al., 2016; Stabile et al., 2017) on the melt viscosity demonstrated that pantelleritic melts are of extremely low viscosity, with respect to their calcalkaline counterparts, regardless of the water content and iron oxidation state. As a consequence of these low viscosities (and hence fast relaxation timescales), the calculated strain rates required for exceeding Maxwell’s criterion and inducing brittle fragmentation in these melts are unrealistically high (Dingwell, 1996; Hughes et al., 2017; Papale, 1999). Furthermore, the crystal content measured in the eruptive products is relatively low; <20 vol% (Campagnola et al., 2016; Gioncada and Landi, 2010; Hughes et al., 2017; Lanzo et al., 2013; Neave et al., 2012; Rotolo et al., 2013). Therefore, the effect of solid particles on the magma viscosity is expected to be very low (Campagnola et al., 2016; Mader et al., 2013 and references therein) and not enough to explain the transition from effusive to explosive eruptions at Pantelleria Island (Civetta et al., 1984; Gioncada and Landi, 2010; Mahood and Hildreth, 1986; Rotolo et al., 2013). It thus appears that the pre-eruptive conditions, including chemical composition and eruptive temperature (Di Carlo et al., 2010; Gioncada and Landi, 2010; Lanzo et al., 2013; Neave et al., 2012; Rotolo et al., 2013; White et al., 2009), are not the decisive factor controlling the intensity of the eruptions occurred at Pantelleria Island. We propose that the increase in viscosity due to nanolite formation induces a rheologic transition pushing the melt across the viscous/brittle transition. This is supported by the presented nano-heterogeneities in melt inclusions of the Green Tuff eruption.

We suggest that future studies should focus on the identification and characterization of nanolites in natural products such as the method presented here or TEM in combination with FE-SEM analysis as presented by Mujin et al. (2017), and on the development of a systematic understanding of the dynamics of their formation in silicate melts at physical and chemical conditions relevant for volcanic eruptions. This will help better constrain the control of melt structure on the eruptive behaviour of volcanoes and on changes in their eruptive style.

6. Conclusions

The presented Raman spectra of synthetic glasses collected with different lasers demonstrate that the spectral fluorescence greatly decreases with decreasing laser wavelength, and blue lasers provide the best signal to background ratio, especially in iron-poor systems.

We further demonstrate that independent of the excitation source, the Raman features at ~670–690 and ~970 cm⁻¹ are exclusively controlled by the presence of magnetite nanolites and changes in the Fe³⁺/Fe⁺ ratio of the glass, respectively. Based on this relationship we develop a new method for the use of Raman spectroscopy as a tool for identification of nanoscale heterogeneities in glasses and introduce the N# as a measure for the degree of nano-heterogeneity of silicate glasses. This approach is significantly faster and more cost-effective than previously used tools such as for example Transmission Electron Microscopy (TEM).

The spectroscopic and analytical results, obtained from melt inclusions and glass embayments of volcanic products erupted during Strombolian (Cuddia del Gallo) and Plinian (Green Tuff) eruptions at Pantelleria Island (Italy), help to unravel the importance of the incipient melt nanolitization for processes acting during explosive eruptions. The acquired Raman spectra show systematic variations along melt inclusions communicating with the surrounding. They document that nanolite occurrence is intimately related to degassing. Since both the precipitation of nanolites and loss of water from the melt result in
increasing magma viscosity, we suggest that nanolite growth in pantelleritic melts contributes to controlling the transition in the eruptive style of such a low viscosity magma.

Based on the data presented above, we propose that glass embayments can be used to study the dynamic evolution of the melt nanostucture during degassing and incipient crystallization prior to and during the eruption. Our contribution illustrates that Raman spectroscopy represents a fundamental tool to investigate the distribution of nanolites, estimate the water content as well as iron oxidation state of volcanic products, and link the nanostucture of silicate melts to the eruptive dynamics.

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Appendix A. Supplementary data

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