Giant photovoltaic response in band engineered ferroelectric perovskite

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Recently the solar energy, an inevitable part of green energy source, has become a mandatory topic in frontier research areas. In this respect, non-centrosymmetric ferroelectric perovskites with open circuit voltage ($V_{OC}$) higher than the bandgap, gain tremendous importance as next generation photovoltaic materials. Here a non-toxic co-doped Ba$_{1-x}$(Bi$_{0.5}$Li$_{0.5}$)$_x$TiO$_3$ ferroelectric system is designed where the dopants influence the band topology in order to enhance the photovoltaic effect. In particular, at the optimal doping concentration ($x_{opt} \sim 0.125$) the sample reveals a remarkably high photogenerated field $E_{OC} = 320$ V/cm ($V_{OC} = 16$ V), highest ever reported in any bulk polycrystalline non-centrosymmetric systems. The band structure, examined through DFT calculations, suggests that the shift current mechanism is key to explain the large enhancement in photovoltaic effect in this family.

The advancement of semiconductor $p$-$n$ junction solar energy in last few decades played a pivotal role as a clean energy resource. However, the open circuit voltage ($V_{OC}$) being lower than the band gap ($E_g$), in these materials, posed obstacle for the future development of $p$-$n$ junction solar cell. Several new materials have been studied for their improved efficiency and photovoltaic (PV) characteristics. Interestingly, ferroelectric oxides are reported to show anomalous PV effect with above bandgap voltage. In fact, Glass et al. observed giant photovoltaic response in Fe doped LiNbO$_3$ single crystal with photogenerated field of the order of $10^4$ V/cm. The origin of such anomalous photovoltaic effect is generally ascribed to ballistic and shift current phenomena. However, the progress of this field demands research on wide range of ferroelectric materials. So far, the PV studies are limited to only few ferroelectrics, namely (Pb,La)(Zr,Ti)O$_3$, BaTiO$_3$ (BTO), Fe:LiNbO$_3$, PbTiO$_3$, (K,Ba)(Nb,Ni)O$_3$, BiFeO$_3$, Pb(Zr,Ti)O$_3$, KBiFe$_2$O$_5$. The observed PV in these ferroelectrics is an outcome of inter-band carrier transitions and hence is largely dependent on the extent of delocalization of valence and conduction states which in turn is determined through the strength of covalent bonding. Among ferroelectric ABX$_3$ perovskites, the tetragonal BTO shows poor PV response due to localized Ti $d$ orbital forming the conduction band edge (CBE) states, whereas CH$_3$NH$_3$PbI$_3$ shows better PV response due to delocalized and spin orbit coupled $p$ states of Pb$^{2+}$ ion in CBE. In fact previous reports show that the shift current response is highly dependent on the orbital character of CBE. If the orbitals forming the CBE has same orientation as that of the polarization direction, there is large shift current which in turn enhances the PV response. For example studies on rhombohedral (K,Ba)(Nb,Ni)O$_3$, where the polarization direction is along the body diagonal, show the degenerate $t_{2g}$ orbitals $(xy, xz$ and $yz)$ at CBE which results in large shift current response. Whereas, for the tetragonal structure with polarization along $z$, the $xy$ orbital forms the CBE leading to a low shift current response.

In this work, we have engineered the band structure of the well-known non-centrosymmetric BTO through doping so that the delocalized CBE is formed by the orbitals oriented along the direction of polarization. Both Bi and Pb are found to be favourable dopants at the A-site and form covalent bonding with the O-$p$ states so that the states in the vicinity of the Fermi level can be made more delocalized. Pb being toxic Bi is preferred as dopant. With Bi$^{3+}$ doping at Ba$^{2+}$ site, a monovalent co-dopant is necessary for charge compensation. In this regards we find Li is an appropriate co-dopant as it is completely ionic and does not affect the frontier electronic structure. We demonstrate that the Bi and Li co-doped BTO samples reveal enhancement in $V_{OC}$ with composition, the highest value being 16 V for the optimal composition $x = 0.125$. 

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Results and Discussion

The Ba$_{1-x}$(Bi$_0.5$Li$_0.5$)$_x$TiO$_3$ (BBLT) samples, $x = 0, 0.05, 0.075, 0.1, 0.125$ and $0.15$, are synthesized by solid-state method. X-ray diffraction (XRD) patterns for $x = 0.125$ (Fig. 1(a)), $0, 0.05, 0.075, 0.1$, and $0.15$ (Supplementary Information Fig. 1(a) to (e)) show pure tetragonal phase with $P4mm$ space group as confirmed by Rietveld refinements. For large doping concentration ($x > 0.125$) the tetragonality is reduced which is illustrated through merging of peaks (Fig. 1(b)) and is further reaffirmed by the ratio of lattice parameters, $c/a$, approaching towards unity (Fig. 1(c)). However, the $c/a$ ratio shows a subtle increment for $x = 0.05$ and thereafter decreasing trend with $x$ (Fig. 1(c)) which can be correlated with tilting of oxygen octahedra associated with expansion of the lattice constant $a$ in the region of $0.075 \leq x \leq 0.15$.

The decrease in cell-volume with $x$ (Fig. 1(c)) is due to the smaller ionic radii of the dopants. The temperature variation of real part of permittivity ($\varepsilon'$) and loss factor (tan$\delta$) at $1$ kHz from $0$ to $200^\circ$C (Fig. 1(e),(f)) unveil the ferroelectric to paraelectric transition ($T_C$) which is at $124^\circ$C for BTO; increased to $137^\circ$C for $x = 0.05$ and thereafter decreased to $129, 108, 79$, and $55^\circ$C for $x = 0.075, 0.10, 0.125$, and $0.15$, respectively. The change in $T_C$ with composition is in accordance with change in $c/a$ ratio (Fig. 1(c)) as reported in several A-site doped BTO samples.$^{24-26}$ Additionally, we find that $x = 0.15$ composition exhibits relaxor behaviour as evidenced from the shift in $T_C$ with frequency (Supplementary Information Fig. 2).

The polarization versus electric field hysteresis loops measured at $4$ Hz (Fig. 2(a)) reiterates the prevailing ferroelectric state of all samples. The remnant polarization ($2P_r$) is enhanced compared to BTO value for each doping concentration.
concentration (Fig. 2(b)). Notably, \(x = 0.10\) composition exhibits \(2P_r = 19.5 \mu \text{C/cm}^2\) which is nearly 100% more than BTO. In fact, our first principle calculations demonstrate that the total ionic dipole moments \(p_i\) follow similar trend with composition (Fig. 2(b)). The samples show nearly invariant optical bandgap of around 3.2 eV, as deduced from Kubelka-Munk plot based on diffused reflectance spectrum (Fig. 2(c))\(^{27,28}\). The optical absorption spectra measured from DFT calculations further confirms the invariance (Fig. 2(c), inset).

To examine the photovoltaic response of the BBLT samples, the adopted capacitor includes a finger geometry as top electrode (Fig. 3(a)) and the current density \((J)\) versus bias voltage \((V)\) is measured under dark and 160 mW/cm\(^2\) light illumination (Fig. 3(b)). Upon co-doping, there is a remarkable enhancement in \(V_{OC}\) (Supplementary Information Fig. 3). In fact, \(x = 0.125\) composition (Fig. 3(b)) exhibits the maximum \(V_{OC}\) of 16 V \((E_{OC} = 320 \text{ V/cm})\) with \(J_{SC} = 9.18 \text{nA/cm}^2\). To the best of our knowledge the displayed \(V_{OC}\) of the present polycrystalline sample \((x = 0.125)\) brings a fold increase in the earlier reported value on single crystalline BTO\(^6\). According to the comparison histogram (Fig. 3(c)) it is indeed the largest ever reported \(V_{OC}\) for any bulk polycrystalline ferroelectric oxides including few single crystalline compounds\(^6,7,16,18,29,30\). Surprisingly, even though \(x = 0.15\) composition has polarization the PV response almost ceases to exist (Supplementary Information Fig. 4(d)). It suggests that there is a simultaneous effect of lattice polarization and conduction band topology to create an optimum composition \((x \approx 0.125)\) for which the PV response is maximum as will be understood from the DFT calculations.

The photocurrent response under zero bias measured in light ON and OFF states for \(x = 0.125\) (Fig. 3b), 0.05, 0.075, 0.10 and 0.15 (Supplementary Information Fig. 4(a) to (d)) samples confirm a quick photocurrent response. The spikes observed are due to the pyroelectric response\(^6\). Notably, the PV response of \(x = 0.125\) sample illustrates a systematic increase in \(J_{SC}\) with light intensity while \(V_{OC}\) remains constant (Fig. 3(d)). Although the light polarization dependent PV response could provide direct evidence for bulk photovoltaic effect (BPVE), the \(V_{OC}\) being independent of light intensity can also indicate the signature of BPVE\(^8,11,31,32\). To investigate the photovoltaic response on polarization state, \(J-V\) measurements and time dependent photocurrent responses are carried out on samples subjected to positive and negative poling fields. The respective plots for \(x = 0.125\) are shown in Fig. 4(a),(b). The plots reveal a sign change on \(V_{OC}\) and \(J_{SC}\) with change in poling state. The observed switchable PV response elucidates the major role of polarization than the electrode contribution.

Since BPVE is largely influenced by the band structure in the vicinity of the Fermi level, we have carried out band structure calculations using DFT (see computational method) to examine the role of the dopants in enhancing the PV response. Bader charge of the ions (Table 1) and deviation of this from the ideal charge state provides a measure of covalency. As the table shows, Ti\(^{4+}\) and O\(^{2-}\) exhibit a significant deviation of \(-1.9\) e and \(0.8\) e...
respectively, to suggest that there are reasonable $p-d$ and $p-p$ covalent interactions. On the other hand for $\text{Ba}^{2+}$, the deviation is small (0.44 e) which suggest that Ba is more ionic. Now with Bi doping, the covalent interaction at the A-site increases significantly as it brings deviation as large as $\sim 1.2$ e.

The covalent interactions are reflected in the band dispersion. In bulk BTO, the lower lying conduction bands (CBM-1 and CBM-2) (Fig. 5, first row) are occupied by the antibonding Ti-$t_{2g}$ states. The polarized BTO, breaks the cubic symmetry to lower the Ti-$xy$ state which now dominates the CBM-1, while $xz$ and $yz$ states occupy CBM-2. However, with doping of Bi and Li, the nature of Ti-O polarization around the dopant neighborhood changes as the TiO$_6$ complex is distorted. This leads to a significant presence of $xz$ and $yz$ orbitals in CBM-1. For $x = 0.125$ (third row, Fig. 5), the CBM-1 and CBM-2 are occupied by all the $t_{2g}$ states. However, for $x = 0.25$ (fourth row, Fig. 5), we find that the CBM is again dominated by the $xy$ states while CBM-2 is by the other $t_{2g}$ characters. Beside Ti-O polarization, the doping brings an additional Bi-O polarization along the $z$-axis and it breaks the three fold degeneracy of Ti-p states. As seen from the Bi-p DOS and charge densities (Fig. 5), the contribution of the $p_z$ state in CBM-1 gradually increases with doping. Additionally, the shape of the charge densities highlights the Bi-O covalent interaction. A larger doping concentration strengthens both Bi-O polarization and Bi-O interaction. As a consequence the antibonding Bi-p states lie above the CBE and do not contribute to PV response. Also we find that large Bi-O polarization decreases the Ti-O bond length along $z$ leaving the $xy$ states lower in energy to form the CBE which further decreases the PV response supporting the experimental results.

According to shift current theory, if the CBE is more occupied by $z$-axis oriented orbitals (e.g. $p_z$, $xz$, $yz$, $z^2-1$) and is very delocalized, resulting from stronger covalent interaction along the polarization direction, the shift current response is high and a large BPVE is observed. As discussed above, in the case of BBLT, the following trends are observed. (I) Initially the polarization increases with $x$ and peaks around $x = x_{opt} (-0.125)$ and decreases afterwards. (II) The covalency increases with $x$ and it is attributed to Bi-$p$ – O-$p$ interactions. (III) The conduction band edge is more occupied by the $z$-axis oriented orbitals (Bi-$p_z$, Ti-$x$-$z$, and yz) and the contribution become maximum around $x = x_{opt}$. Therefore, based on shift current theory, the BPVE should increase with $x$ and become maximum at $x_{opt}$. We may note that since the calculations are carried out for some discrete values of $x$, exact value of $x_{opt}$ cannot be determined theoretically. However, experimentally we find that the BPVE is maximum at $x = 0.125$.

**Conclusion**

In summary, we have demonstrated a unique way to tailor the bands via A-site doping to create giant photovoltaic response in polycrystalline perovskite ferroelectric oxides. In the present work, this has been achieved by co-doping Bi and Li in BaTiO$_3$. This work opens up an opportunity to design new family of ferroelectric...
compounds for photovoltaic studies by carefully choosing the dopants in such a way that it will lead to simultaneous increase in lattice polarization and delocalization of the conduction band edge state.

Materials and Methods

Experimental section. \( \text{Ba}_1-x(\text{Bi}_x\text{Li}_{1/2})_x\text{TiO}_3 \) with composition \( x = 0.0, 0.05, 0.075, 0.1, 0.125 \) and 0.15 were fabricated by solid state method. Stoichiometric mixture of initial precursors \( \text{BaCO}_3 \) (purity \( \geq 99.9\% \)), \( \text{TiO}_2 \) (purity \( \geq 99.9\% \)), \( \text{Bi}_2\text{O}_3 \) (purity \( \geq 99.9\% \)) and \( \text{Li}_2\text{CO}_3 \) (purity \( \geq 99.9\% \)) were ground in an agate mortar and calcined at 700 °C for 4 h followed by 800 °C for 6 h. The uniaxially pressed pellets of 12 mm diameter made from calcined powder of various composition were fired at optimized temperatures in the range of 1000 to 1300 °C for 2 h. The powdered samples were subjected to X-ray diffraction (Rigaku smartLab) experiments for phase confirmation. The Rietveld refinement was done on the XRD pattern by FULLPROF software. For dielectric and polarization measurements, the 12 mm diameter sintered pellets were coated with Ag as electrodes. The temperature dependent dielectric permittivity and loss factor were obtained from Novocontrol impedance analyser in 10 Hz to 10 MHz range. The polarization measurements were carried out by employing Radiant Technology loop tracer at 27 °C. The Ultraviolet–Visible–Near Infrared (UV–VIS–NIR) Spectro-Photometer (JASCO V-650) was employed for optical band gap measurements. For PV measurements, the polished samples were cut into 7 × 7 mm square and 0.5 mm thickness dimension. On the top side of the sample, silver electrodes of size 6.4 × 6.4 mm (finger thickness 100 μm and gap 200 μm) were deposited by thermal evaporation technique using shadow mask and the bottom side is covered by Ag paint. The samples were poled by 20 kV/cm field in pressure contact mode over entire electrode area using a thin copper plate of 0.15 mm thickness on the top finger electrode. The effective poling was confirmed by comparing the \( d_{33} \) values with and without finger electrode. Upward/downward poling implies the application of a negative/positive voltage to the top electrode. PV measurements were carried out by employing Xenon-arc lamp (Newport, Model No-67005) as light source and Keithley electrometer (6517B).
as measuring unit. The values of current density, \( J_{SC} \), were arrived by considering the entire top surface area (7 × 7 mm²) in the calculation.

**Computational method.** Present DFT calculations are performed using Pseudopotential (PP) based Vienna *ab-initio* Simulation Package (VASP)\(^{34} \). PPs are based on projected augmented wave (PAW) method with exchange and correlation effects described using Generalised Gradient Approximation (GGA) and Perdew-Burke-Ernzerhof (PBE) functional\(^{35,36} \). For the elements present in our model, the following valence electrons were explicitly considered in the PP: Ba: 4s\(^2\) 5p\(^6\) 5s\(^2\), Li: 1s\(^1\) 2s\(^1\), Bi: 6p\(^3\) 5d\(^4\) 6s\(^2\) 6p\(^3\), Ti: 3p\(^2\) 3d\(^4\) 4s\(^2\) and O: 2s\(^2\) 2p\(^4\). In order to account for relativistic effects arising from heavy Bi element, spin orbit coupling (SOC) has been included in our calculations. SOC calculations were performed as implemented in PAW methodology in VASP package\(^{37,38} \). Since SOC is predominantly act in the immediate vicinity of nuclei, the SOC Hamiltonian is solved self consistently along with PAW Hamiltonian within the PAW sphere. The plane wave cut-off energy was chosen as 500 eV. To simulate experimental doping concentrations we adopted supercell approach with periodic boundary conditions and performed our calculations using \( 3 \times 3 \times 3, 3 \times 3 \times 2, 2 \times 2 \times 4 \) and \( 2 \times 2 \times 2 \) supercells corresponding to \( x = 0.07, 0.11, 0.125, \) and 0.25 respectively. In each of these supercells, two Ba ions are replaced by single Bi and Li ions. We performed our calculations with Li and Bi in four different orientations [001], [100], [110], and [111] corresponding to Bi and Li bond lengths 4.03 Å, 3.98 Å, 5.63 Å, 6.94 Å respectively. The total energy and band structure of these configurations are found to be nearly identical. Therefore, in the present work only one of them [001] is presented in detail. For structural optimization, we chose a convergence criterion of \( 1 \times 10^{-6} \) eV for self-consistent field (SCF) electronic energy and \( 1 \times 10^{-3} \) eV/Å for Hellmann–Feynman forces on each atom. A \( 8 \times 8 \times 8 \) Monkhorst–Pack grid is used to sample the Brillouin–Zone integrations of bulk BaTiO\(_3\).\(^{39,40} \) Proportionate \( k \)-grids are used for the supercells. The total ionic charges were calculated using the Bader Atom Molecule (AIM) approach as implemented in the program by Arnaldsson et al\(^{39,40} \).

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**Author Contributions**
S.P., A.B.S. and P.M. planned the material design. S.P. and A.B.S. and P.P.B. synthesized the materials. P.M. designed the experiments and supervised the obtained results. S.P. and A.B.S. carried out the XRD measurements. A.P. performed the Rietveld analysis on XRD data. S.P. carried out the dielectric and optical measurements. A.B.S. performed the polarization measurements. S.P., A.B.S. and P.P.B. performed the photovoltaic measurements. S.P., A.B.S., P.P.B. and D.M. equally contributed for this work. D.M. and B.R.K.N. designed and performed the DFT calculations. B.R.K.N. and P.M. co-wrote the paper with input from all the contributing authors.

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