Arrangements of nanostructures in well-defined patterns are the basis of photonic crystals, metamaterials and holograms. Furthermore, rewritable optical materials can be achieved by dynamically manipulating nanoassemblies. Here we demonstrate a mechanism to configure plasmonic nanoparticles (NPs) in polymer media using nanosecond laser pulses. The mechanism relies on optical forces produced by the interference of laser beams, which allow NPs to migrate to lower-energy configurations. The resulting NP arrangements are stable without any external energy source, but erasable and rewritable by additional recording pulses. We demonstrate reconfigurable optical elements including multilayer Bragg diffraction gratings, volumetric photonic crystals and lenses, as well as dynamic holograms of three-dimensional virtual objects. We aim to expand the applications of optical forces, which have been mostly restricted to optical tweezers. Holographic assemblies of nanoparticles will allow a new generation of programmable composites for tunable metamaterials, data storage devices, sensors and displays.
Programmable materials that change their physical properties are a major topic of interest in modern science. Mechanisms to configure nanostructures in three-dimensional (3D) space are essential in nanotechnology, photonics and materials science. Common optical nanofabrication methods rely on light-sensitive materials such as silver halides and photoresists. Highly intense laser pulses can ablate materials with spatial selectivity, altering their physical and optical properties. Various nanopatterning techniques based on photoactive materials or photoablation have been used to produce static photonic crystals, lasers, metamaterials, holograms, storage devices and sensors. Optically rewritable materials have been demonstrated with photorefractive or photochromic media by changing their refractive indexes locally. However, they require external energy to maintain the information. Although some dynamic mechanisms have been applied at nanoscale to assemble crystal structures and twisted ribbons, the dynamic displacement of 3D nanoassemblies to form well-defined configurations inside solids is not easily accomplished.

Dielectric and metal nanoparticles (NPs) in viscoelastic media have a complex behaviour in the presence of radiation gradients. An optical force (tractor force) results from the momentum transfer associated with the spatially asymmetric light scattering and absorption of a nanostructure. Electromagnetic forces in gradients can push particles towards regions of maximum intensity (positive forces) or minimum intensity (negative forces). In dielectrics, the force can be positive or negative when the NP has higher or lower refractive indexes than the medium respectively. The phase shift of the scattering dictates the direction of the force. In metal NPs, the phase and intensity of the scattering depends on the surface plasmon resonance produced by the free electron cloud. Hence, the direction of the optical force is dictated by different factors including geometry, size and material of the NP, the surrounding medium and the wavelength of the applied field. An assembly of NPs embedded in a solid can be reconfigurable with optical forces when the viscoelasticity of the medium permits the migration and the stabilization in a reversible manner. When the optical force passes a threshold, NPs overcome surface adhesion, elastic forces and the static friction induced by the medium.

Here we introduce a strategy based on non-ablative optical pressure to arrange nanostructures inside transparent solids with viscoelastic characteristics. The shape-memory characteristics of some types of polymers provide the necessary plasticity for reversible nanomaterial configuration in situ. We use optical standing waves to control heat and optical force to arrange NPs in different 3D configurations. In addition to the optical force, thermodynamic (thermophoresis) and acoustic (acoustophoresis) forces arise due to mechanical pressure produced by temperature increase in the medium, which pushes NPs towards the minimum light intensity regions. However, low dissipation of temperature and mechanical pressure are necessary to achieve thermophoresis and acoustophoresis. Although these effects can produce reconfigurability, the system presented in this work is in the optical force regime (Supplementary Notes 1 and 2, and Supplementary Fig. 1). To our knowledge, this optical force-induced mechanism to assemble nanostructures in organized, reversible configurations in solids has not been reported previously. Using this mechanism in the negative force regime, we demonstrate rewritable photonic crystals, optical elements and 3D holograms.

**Figure 1** | Mechanism of nanostructure reconfiguration in a medium. (a) Schematic of the nanoassembly process, which consist of the following: an initial state where NPs are randomly distributed, a transition state where NPs migrate to lower-energy configurations and a final assembly state where NPs are located in new stable positions (Supplementary Movie 1). (b) Rheology of a typical thermoplastic (for example, the complex shear modulus $(G' + i G'')$ of pHEMA changes from $1.4 \times 10^4 + i 2.0 \times 10^4$ Pa in the glassy regime to $2.9 \times 10^4 + i 2.0 \times 10^4$ Pa well above its glass transition temperature $(T_g)$ of 300°C). (c) Time-dependent temperature at the surface of NPs of different diameters from a pulsed laser 532 nm, 5 ns and 20 mJ cm$^{-2}$ (this is equivalent to the temperature produced by two interfering waves of 10 mJ cm$^{-2}$ at the maximum gradient point). (d) Dynamics of NP displacement in a standing wave. (e) Relative maximum force acting on Ag NPs of different diameters with a standing wave of 532 nm. (f) Potential wells produced by the interference of counter-propagating waves.
Results

Migration of NPs in standing waves. Metal NPs can absorb optical energy and allow the converted thermal energy to dissipate to their surrounding medium. Hence, the medium behaves as a fluid locally, allowing NPs to be displaced from their original positions. Finally, the composite returns to its original glass state after the displacement (Fig. 1a and Supplementary Movie 1). The analysis of temperature at the NP and medium is necessary to prevent transition of phase. We consider Ag NPs for their high optical scattering and absorption, and poly(2-hydroxyethyl methacrylate) (pHEMA) as the embedding medium for its unique rheological characteristics. The phase transition temperature of Ag NPs is lower than bulk Ag but higher than the degradation temperature of pHEMA, both of which are slightly above 300 °C.34,35. Furthermore, the temperature at the boundary of the NPs dictates the mechanical properties of the surrounding medium. As pHEMA has low heat conduction, the high temperature at the NP–pHEMA boundary allows the pHEMA matrix to behave similar to a viscoelastic rubber. This phenomenon is analogous to ‘the knife in the butter’, where the medium changes its stiffness according to the temperature of the metal. Furthermore, this effect is present as long as the heat of the metal diffuses in the pHEMA matrix. We have rationally designed a pHEMA matrix that can reversibly transform from its glass state to its rubbery state by increasing the temperature at the NP boundaries (Fig. 1b). A numerical analysis shows the temperature at the boundaries of spherical Ag NPs when they are heated by a Gaussian laser pulse (Fig. 1c). Alternatively, the temperature at the surface $T_{\text{CW}}(t)$ in time can be approximated for a continuous irradiance $I_{\text{CW}}$ as (Supplementary Note 3 and Supplementary Fig. 2):

$$T_{\text{CW}}(t) \approx I_{\text{CW}} C_{\text{abs}} \left( \frac{r}{4K_{\text{m}}} \right) \left( 1 - \exp\left( -\frac{3K_{\text{m}} z_{\text{NP}}}{K_{\text{NP}} r^2 t} \right) \right)$$

where $C_{\text{abs}}$ is the absorption efficiency, $K_{\text{NP}}$ and $z_{\text{NP}}$ are the thermal conductivity and diffusivity of the NP, respectively, and $K_{\text{m}}$ is the thermal conductivity of the medium. The optical forces $F_p$ can invert the direction of the force for a wavelength near the surface plasmon resonance. Furthermore, forces in metal NPs can be up to one order of magnitude higher than their dielectric counterpart. An approach to retrieve the force below the first plasmonic resonant mode is with the Lorentz–Lorenz equation:

$$F_p = \frac{\Re(\chi)}{2c} \nabla I$$

where $c$ is the speed of light and $\Re(\chi)$ represents the real part of the complex polarizability that can be obtained as:

$$\chi = \frac{4}{3} \pi \varepsilon_0 \varepsilon \left( \frac{n_p^2 - n_m^2}{n_p^2 + 2n_m^2} \right)$$

where $n_p$ and $n_m$ are complex refractive indexes of a NP and surrounding medium, respectively. This approximation is valid for a Ag NP radius smaller than ~35 nm. A further effective change in sign can occur for NPs with a size comparable to the period of the standing wave, because the two neighbouring nodes have opposite direction in the force.36. Depending on the size, NPs settle at the maximum intensity or minimum intensity regions of the interference fringe.37. A more accurate approximation for metal NPs can be retrieved from the generalized Lorenz–Mie theory (GLMT). In GLMT, the electromagnetic field consisting of incident and scattered field outside the sphere, and the internal field within the sphere, is expanded into a series of multipole terms.39

We define arbitrary standing waves with the interference of two coherent beams propagating normal to the plane $z$ with an included angle $\theta$ and intensity $I_0$. The standing wave created is $I = I_0 \sin(\theta/2)\left(1 + \sin(k_z z + \phi)\right)$, with a wavenumber $k_z = 2\pi/A$, a grating spacing $A$ and a phase $\phi$. Furthermore, the phase of the standing wave is controlled with the relative phase difference between the beams, whereas the grating spacing is defined as $\lambda = \lambda/(2n_{\text{eff}} \sin(\theta/2))$, for a given beam wavelength $\lambda$ and a medium effective refractive index of $n_{\text{eff}}$. Hence, the force exerted by the standing wave of two counter-propagating beams ($\theta = 180^\circ$) is $F_p \propto \cos(4\pi z/\lambda + \phi)$. According to the Fourier theorem, an arbitrary intensity profile can be obtained from the superposition of waves.38 Figure 1d shows the displacement of NPs from bright regions to dark regions in a standing wave produced by a negative force. To retrieve the migration, we applied the generalized Stokes’ law for a NP of radius $r$ embedded in a complex viscoelastic medium with a shear modulus $G = 2.9 \times 10^4 + 12.0 \times 10^4$ (Supplementary Note 1):39

$$z_{\text{m}}(\omega) = \frac{F_{\text{m}}(\omega)}{6\pi r G(\omega)}$$

where $z_{\text{m}}(\omega)$ is the complex NP displacement and $F_{\text{m}}(\omega)$ is the complex force (representing the amplitude and phase of the oscillation). The total migration close to the maximum gradient point produced by the interference of two counter-propagating waves is approximated as:

$$\Delta z \approx \frac{1.25 F_{\text{max}}}{rG} U$$

where $k$ is the wave number, $U$ is the total exposure energy, $r_p$ is the full width at half maximum pulse size and $F_{\text{max}}$ is constant that satisfies: $F_p = F_{\text{max}} \nabla I$, which can be obtained either from the Lorentz–Lorenz or Lorenz–Mie equations.

Figure 1e shows the relative maximum force of Ag NPs considering 532 nm light source for the Lorentz–Lorenz and the Lorenz–Mie equations. The potential energy $V$ is proportional to the intensity of the wave $V \propto I_0$, as the force correlates with the gradient of the intensity. Figure 1f illustrates the energy potential produced and forces exerted by the interference of four arbitrary waves. The potential wells represent equilibrium positions, because the velocity of NPs is damped by the viscosity of the medium.

Nanoassembly in composites. We calculated the optical force induced by a 532-nm laser pulse of 5 ns and 10 mJ over 1 cm² with the GLMT applied to Ag NPs (Fig. 2a). Figure 2b shows the NP displacement with respect to the total exposure retrieved from Stokes’ law (Supplementary Note 1). The maximum force and displacement was observed for NP sizes in the 30–90 nm range. Based on these values, we optimized the fabrication for Ag NPs in the 30–90 nm range embedded in a pHEMA film (~10 μm in thickness) (see ‘Fabrication of the Recording Media’ in Methods). Figure 2c illustrates the optimized size distribution of Ag NPs embedded in the pHEMA matrix. To form the holographic reconfiguration, we used a Nd:YAG (532 nm, 5 ns) pulsed laser. We measured the diffraction efficiency of the recorded multilayer Bragg grating with respect to the incident beam intensity. An interference was produced from two counter-propagating beams in Denisyuk reflection mode with a tilt angle of 5° with respect to the surface plane of the pHEMA matrix. When the radiant fluence was below 1 mJ cm⁻², no recording effect was observed. However, when the fluence was >20 mJ cm⁻², the absorption at the NP surface degraded the pHEMA matrix. The optimum
fluence was 10 mJ cm\(^{-2}\), to maximize the diffraction efficiency. This fluence produced a maximum temperature of 300 °C at the NP surfaces, corresponding to the limit where pHEMA degraded\(^{35}\). The NP displacement was increased by repeating the number of pulses without compromising the integrity of the pHEMA matrix. We validated the migration by measuring the diffraction efficiency from the multilayer Bragg grating recorded in Denisyuk reflection mode (see ‘Refractive index measurement’ in Methods). Ag NPs were arranged in a slanted 3D structure with a periodicity of \(\sim \lambda/2\). Figure 2d shows the increment in the diffraction efficiency reaching a maximum of 2.4% as the number of pulses increased up to 200 exposures (at 10 Hz). This behaviour is similar to a damped harmonic oscillator, where the dragging force settles the NPs in a steady position. The theoretical curve obtained (red line) agreed with the measured values (points) and the fitted curve (blue dashed line). Based on these parameters, the measured sensitivity of the fabricated composite was 885 mJ cm\(^{-2}\) to reach the 90% maximum diffraction efficiency.

We recorded multilayer structures by tilting the sample at different angles with respect to the standing wave. The fabricated gratings corresponded to the slanted cross-section of a multilayer structure. We illuminated the Bragg grating at normal incidence with a supercontinuum white light laser. The backscatter light was measured at different angles using a spectrophotometer positioned onto a goniometer (Fig. 3a). The band gap of the photonic structure had a peak at \(\sim 527\) nm and a full width at half maximum bandwidth \(\Delta \lambda\) of 36 nm. The 5-nm blue shift with respect to the recording laser was consistent in different trials. The blue shift could be attributed to a temperature difference during recording and readouts. During recording, the composite slightly expanded because of heat, whereas during readout the sample was at room temperature. An approximation of the recorded multilayer length can be extracted by considering a ‘weak-grating’ approximation \(L \approx \frac{\lambda}{2\Delta n_{\text{eff}}}\) (‘Refractive index measurement’ in Methods). The resulting length of the multilayer Bragg grating is \(\sim 5.39\) µm corresponding to \(\sim 29\) bilayers (Fig. 3b). In addition, the modulation in the effective refractive index \(\Delta n_{\text{eff}}\) through the expression is \(\Delta n_{\text{eff}} \approx \frac{\Delta \lambda}{2\lambda}\) for a given diffraction efficiency of \(R\). The estimated change in refractive index between fringes \(2\Delta n_{\text{eff}}\) is \(\sim 0.01\). The fabricated nanostucture served as a narrow-band wavelength-selective filter to diffract an intense colour at 8° away from the sample normal. Figure 3c shows an optical microscope image of a Ag NP transmission grating in the pHEMA matrix, consisting of a sinusoidal pattern (\(\sim 3\) µm).

**Rewritable nanophotonic devices.** To demonstrate reversibility, we recorded a grating at 5° from the surface plane and erased it (recorded at 0°) iteratively for 12 times (Fig. 4a, Supplementary Note 4 and Supplementary Fig. 3). The holographic patterning technique can be used to configure different crystal plane orientations (Supplementary Note 5 and Supplementary Fig. 4). Bragg gratings were superposed at 5°, 10°, 15°, 20° and 25° to form photonic crystals (Fig. 4b). To erase the pattern, pHEMA matrix was aligned parallel to the surface plane of the object (that is, front-surface mirror) at 0°. This configuration aligned the multilayer structure with the specular reflection (zero order). Figure 4c shows the diffraction of the five band gaps projected at a semitransparent screen.
Crystal structures in binary configurations were also recorded and erased, to demonstrate volumetric data storage (Fig. 4d). The angle-resolved diffraction intensity of the crystal structures were measured with a 532-nm laser (Fig. 4e,f corresponding to the structures in Fig. 4b,c, respectively). The diffracted rays diverged for large angles, because the grating structures embedded in pHEMA matrix had a high refractive index (\( n > 1.43 \)). Figure 4g shows a comparison of the different recorded angles and their theoretical predictions. The diffraction efficiency of the photonic structure decreased as the number of superposed spots increased. When NPs were distributed in multiple fringe planes, the contrast of the effective refractive index decreased.

We also used the optical forces to fabricate dynamic lenses. A concave mirror with a 10-mm focal point was recorded. The focusing of a beam produced by the holographic lens is demonstrated with the superposition of multiple images of the beam path (Fig. 5a). Figure 5b shows a plot of the beam waist through the optical path. In addition, we obtained 3D holographic reconstructions of coins using multiple pulses at 10 Hz by recording a single object within 5 s. On white light illumination, a 3D holographic reconstruction was observed by the eye. The 3D reconstruction exhibited uniform monochromatic brightness. The holographic pattern was completely erased using the same laser beam within 30 s (Fig. 5c). Figure 5d,e shows different coins recorded within the same media after an erasing step.
We also demonstrate image multiplexing capability by superposing two coins (Fig. 5f). By changing the number of pulses and intensity, the object beam can be finely tuned to partially or fully erase a 3D virtual image, or superpose a new hologram.

**Discussion**

To date, the theoretical and empirical demonstrations of optical forces on dielectric or plasmonic particles have been mostly restricted to optical tweezers for the manipulation of NPs in liquids. We presented a mechanism to assemble volumetric optical elements, displays, biosensors, data storage devices and metamaterials.

**Methods**

**Fabrication of the recording media.** 3-(Trimethoxysilyl)propyl methacrylate in acetone (1:50, v/v) was poured onto glass microscope slides in an aluminum tray. After thorough coating, the excessive silane was poured off, while slides remaining dark, before removal and dark storage at 24 °C. A monomer solution consisting of 2-hydroxyethyl methacrylate (91.5 mol%), ethylene dimethacrylate (2.5 mol%) and methacylic acid (6 mol%) was prepared. The monomer solution was mixed with 1:1 (v/v) with photo-initiator 2,2-dimethoxy-2-phenylacetophenone in propan-2-ol (2%, w/v). A few silica beads (10 μm) were added to the solution. The monomer solution (200 μl) was deposited on an aluminized sheet and the glass slide, silanized treated side down, was placed on the solution to allow the capillary force to spread the solution throughout the surface of the glass slide, whereas the silica beads controlled the thickness of the monomer solution. The monomer mixture was cured using ultraviolet light-induced free radical polymerization for 1 h. The pHEMA matrix was rinsed with ethanol (100 vol%) to remove the unreacted compounds from the pHEMA matrix. AgNO₃ solution (0.1 M, 200 ml) dissolved in deionized water (DI) was perfused into the pHEMA matrix for 1 min. The excess salt solution was removed with a squeegee and the film was dried under tepid air current for 5 s. With the pHEMA matrix side facing up, the slide was submerged in a Petri dish containing LiBr (0.3 M, 40 ml) in methanol:water (3:2, v/v) for 30 s, to allow the reduction of AgBr nanocrystals to metal Ag NPs. A photographic developer (JD-4) consisting of 4-methylaminophenol sulfate (0.3%, w/v), ascorbic acid (2%, w/v), Na₂CO₃ (5%, w/v) and NaOH (1.5%, w/v) was prepared. The pH of the developer (pH = 13) for 30 s, to allow the reduction of AgBr nanocrystals to metal Ag NPs. The pH of the developer was washed thoroughly with DI water and immersed into a stop bath consisting of acetic acid (5%, v/v, pH ~ 2.5), to neutralize the developer, for 10 s. The pHEMA matrix was submerged in a Na₂S₂O₃ (10%, w/v) (hypo/X-ray fixer) solution for 5 min.
solution to remove the unreacted AgBr nanocrystals. The images were recorded in the pHEMA matrix in ambient humidity (60% relative humidity (RH)).

**Recording in Denisyuk reflection mode.** The standing waves were formed via in-line Denisyuk reflection mode.\(^1\) Figure 6 illustrates the experimental setup schematic. Laser pulses were directed onto a pHEMA recording medium and reflected from the recording object, which was placed over a translational stage and positioned normal to the laser incidence pulse having a spot size of ~1 cm in diameter (Fig. 6a). The semi-transparent composite is placed between the reference beam and the object reflection beam. In the case of the volume grating, the medium was tilted an angle from the surface mirror plane. The laser pulse energy at the sample-air interface was measured as ~10 mJ using an optical powermeter. The incident pulse propagated through the substrate and the pHEMA recording medium, and reflected off from an object. The interference of the object beam and the reference beam created a standing wave. The recorded object was reconstructed with white light without affecting the assembly (Fig. 6b). The assembly was arranged in multi-layer planes parallel to the surface interface to erase the reconstruction (Fig. 6c). Figure 6d shows a 3D reconstruction observed from a sample recorded in Denisyuk reflection mode.

**Refractive index measurement.** The refractive index of the pHEMA was measured by an Abbe refractometer. The measured refractive index of pHEMA matrix was 1.5061. The pHEMA matrices were lifted off from the sample using an Abbe refractometer. The pHEMA matrices were lifted off from the sample using a plano-convex lens (laser power 25 cm diameter × 75 mm, unfocused) was purchased from Edmund Optics Ltd (York, UK). AvaSoft (v7.5) software, Igor Pro, COMSOL Multiphysics (v5.1) and MATLAB (MathWorks, v8.1) were used for data processing and finite element simulations. DEymond Histo 5/6/8 mm micrometre diamond knife was purchased from Scien GmbH (Am Mühlenkanal, Germany) and a low-temperature sectioning system (microtome) Ultracut Uc was purchased from Leica (Vienna, Austria). NP size distributions were analysed using ImageJ (v1.48p, 64 bit, National Institutes of Health, USA).

**Data availability.** The data that support the findings of this study are available from the corresponding author upon request.

**References**

1. McEvoy, M. A. & Correll, N. Materials that couple sensing, actuation, computation, and communication. *Science* 347, 1328 (2015).

2. Brongersma, H. L. Silver-Holographic Recording Materials: For Holography and Their Processing (Springer, 1995).

3. Mafune, F., Kohno, J., Takeda, Y., Kondow, T. & Sawabe, H. Formation and size control of silver nanoparticles by laser ablation in aqueous solution. *J. Phys. Chem. B* 104, 9111–9117 (2000).

4. Blanco, A. et al. Large-scale synthesis of a silicon photonic crystal with a complete three-dimensional bandgap near 1.5 micrometres. *Nature* 405, 437–440 (2000).

5. Painter, O. et al. Two-dimensional photonic band-gap defect mode laser. *Science* 284, 1819–1821 (1999).

6. Valentine, J. et al. Three-dimensional optical metamaterial with a negative refractive index. *Nature* 455, 376–379 (2008).

7. Larouche, S., Tsai, Y. J., Tyler, T., Jokester, N. M. & Smith, D. R. Infrared metamaterial phase holograms. *Nat. Mater.* 11, 450–454 (2012).

8. Zijlstra, P., Chon, W. M. & Gu, M. Five-dimensional optical recording mediated by surface plasmons in gold nanorods. *Nature* 459, 410–413 (2009).

9. Yetisen, A. K. et al. Reusable, robust, and accurate laser-generated photonic metamaterials. *Nano Lett.* 14, 3587–3593 (2014).

10. Blanche, P.-A. et al. Holographic three-dimensional telepresentation using large area photorefractive polymer. *Nature* 468, 80–83 (2010).

11. Tay, S. et al. An updatable holographic three-dimensional display. *Nature* 451, 694–698 (2008).

12. Killat, U. & Terrell, D. R. Performance and limitations of photothermoplastic devices. *Opt. Acta Int. J. Opt.* 24, 441–452 (1977).

13. Drolet, J.-P., Chuzan, E., Barbabatsis, G. & Patsis, D. Compact, integrated dynamic holographic memory with refreshed holograms. *Opt. Lett.* 22, 552–554 (1997).

14. Gross, R. B., Izgi, K. C. & Birge, R. R. In proceedings for SPIE 1662, *Image Storage and Retrieval Systems 1662*, 186–196 (1992).

15. Klein, R., Bishop, K. J. M. & Grzybowski, B. A. Light-controlled self-assembly of reversible and irreversible nanoparticle suprastructures. *Proc. Natl Acad. Sci. USA* 104, 10305–10309 (2007).

16. Kim, Y., Shah, A. A. & Solomon, M. J. Spatially and temporally reconfigurable assembly of colloidal crystals. *Nat. Commun.* 5, 3676 (2014).

17. Srivastava, S. et al. Light-controlled self-assembly of semiconductor nanoparticles into twisted ribbons. *Science* 327, 1355–1359 (2010).

18. Brezobohaty, O. et al. Experimental demonstration of optical transport, sorting and self-arrangement using a tractor beam. *Nat. Photonics* 7, 123–127 (2013).

19. Siler, M., Chvalat, L. & Zemaneck, P. Metallic nanoparticles in a standing wave: optical force and heating. *J. Quant. Spectrosc. Radiat. Transf.* 126, 84–90 (2013).

20. Manman, K. C. & Block, S. M. Optical trapping. *Rev. Sci. Instrum.* 75, 2787–2804 (2004).

21. Barnes, W. L., Dereux, A. & Ebbesen, T. W. Surface plasmon subwavelength optics. *Nature* 424, 824–830 (2003).

22. Yu, N. et al. Light propagation with phase discontinuities: generalized laws of reflection and refraction. *Science* 334, 333–337 (2011).

23. Chen, W.-T. et al. High-efficiency broadband meta-hologram with polarization-controlled dual images. *Nano Lett.* 14, 225–230 (2014).

24. Dogariu, A., Sukhov, S. & Säenz, J. Optically induced ‘negative forces’. *Nat. Photonics* 7, 24–27 (2013).

25. Lee, K.-S. & El-Sayed, M. A. Dependence of the enhanced optical scattering efficiency relative to that of absorption for gold metal nanorods on aspect ratio, size, end-cap shape, and medium refractive index. *J. Phys. Chem. B* 109, 20331–20338 (2005).

26. Liu, C., Qin, H. & Matther, P. T. Review of progress in shape-memory polymers. *J. Mater. Chem.* 17, 1543–1558 (2007).

27. Mewis, J. & Wagner, N. J. Thixotropy. *Adv. Colloid Interface Sci.* 147–148, 214–227 (2009).

28. Guvendiren, M., Lu, H. D. & Burdick, J. A. Shear-thinning hydrogels for biomedical applications. *Soft Matter* 8, 260–272 (2012).

29. Duhr, S. & Braun, D. Why molecules move along a temperature gradient. *Proc. Natl Acad. Sci. USA* 103, 19678–19682 (2006).

30. Foresti, D., Nabiavi, M., Klingauf, M., Ferrari, A. & Poulikakos, D. Acoustophoretic contactless transport and handling of matter in air. *Proc. Natl Acad. Sci. USA* 110, 12549–12554 (2013).

31. Brenner, H. & Bielenberg, J. R. A continuum approach to photoretic motions: thermophoresis. *Phys. Stat. Mech. Appl.* 104, 19678–19682 (2006).

32. Schermer, R. T., Olson, C. C., Coleman, J. P. & Bucholtz, F. Laser-induced thermophoretic motion of individual particles in a viscous liquid. *Opt. Express* 19, 10571 (2011).

33. Dysthe, K. B. Force on a small inclusion in a standing acoustic wave. *J. Sound Vib.* 10, 331–339 (1969).
34. Little, S. A., Begou, T., Collins, R. W. & Marsillac, S. Optical detection of melting point depression for silver nanoparticles via in situ real time spectroscopic ellipsometry. *Appl. Phys. Lett.* **100**, 51107 (2012).
35. Coşkun, M., Demirelli, K., Erol, I. & Ahmediade, M. Thermal degradation of poly[2-(3-aryl-3-methylcyclobutyl)-2-hydroxyethyl methacrylate]. *Polym. Degrad. Stab.* **61**, 493–497 (1998).
36. Zemanek, P. *et al.* Theoretical comparison of optical traps created by standing wave and single beam. *Opt. Commun.* **220**, 401–412 (2003).
37. Zemanek, P., Jonaš, A. & Liika, M. Simplified description of optical forces acting on a nanoparticle in the Gaussian standing wave. *J. Opt. Soc. Am. A* **19**, 1025–1034 (2002).
38. Goodman, J. W. *Introduction to Fourier Optics* (Roberts & Company Publishers, 2004).
39. Gittes, F., Schurr, B., Olmsted, P. D., MacKintosh, F. C. & Schmidt, C. F. Microscopic viscoelasticity: shear moduli of soft materials determined from thermal fluctuations. *Phys. Rev. Lett.* **79**, 3286–3289 (1997).
40. Yetisen, A. K. *et al.* Mechanism of multiple grating formation in high-energy recording of holographic sensors. *Appl. Phys. Lett.* **105**, 261106 (2014).
41. Benton, S. A. & Bove, V. M. in *Holographic Imaging* 173–180 (John Wiley & Sons, Inc., 2008).
42. Erdogan, T. Fiber grating spectra. *J. Light Technol.* **15**, 1277–1294 (1997).

**Acknowledgements**
We thank Lukas Chvatal, Jeff Blyth and Christopher R. Lowe for discussions.

**Author contributions**
A.K.Y. and Y.M. conceived the idea and performed the experiments. Y.M. developed the theory and carried out the simulations. Y.M. and A.K.Y. wrote the article. A.Y. and H.B. made intellectual contributions and revised the article.

**Additional information**

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Montelongo, Y. *et al.* Reconfigurable optical assembly of nanostructures. *Nat. Commun.* **7**:12002 doi: 10.1038/ncomms12002 (2016).

This work is licensed under a Creative Commons Attribution 4.0 International License. The images or other third party material in this article are included in the article’s Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder to reproduce the material. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/