Properties in Fe-Doped ZnS Thin Films

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Abstract

Fe-doped ZnS single-phase thin films showing ferromagnetism have been successfully prepared by Chemical Bath deposition (CBD) on GaAs substrates. Field and temperature dependent magnetization curve indicate that the sample with a Curie temperature $T_c$ as high as 270K. The X-ray diffraction and Atomic Force Microscopy (AFM) reveal that the thin films are well crystallized, Fe ions are substituted for Zn ions in the ZnS matrix and no trace of secondary phases or Fe clusters is detected. The experimental results are explained theoretically by spin-polarized density functional calculations within Generalized-Gradient Approximations (GGA), which indicates the observed high $T_c$ could be mainly ascribed to the p-d exchange coupling between Fe ions and host elements.

Keywords: Fe-doped ZnS; First principles calculation; Ferromagnetic properties

Introduction

Transition-Metal (TM) doped wide gap II-VI semiconductors materials, such as ZnO, GaN and ZnS, have been extensively investigated for potential magnetoelectronic or magneto-optical applications with the purpose of realizing of a high Curie temperature ($T_c$) [1-3]. Such TM-doped semiconductors called “Dilute Magnetic Semiconductor” (DMS) could be integrated with conventional semiconductor devices for spintronic applications [4-6]. In past years, many works have studied the doping properties of ZnS. A band structure model, based on the p-d and d-d level repulsions between the TM ions and host elements, has been successfully used to explain the magnetic ordering observed in all Mn-doped III-V and II-VI semiconductors. In this model, the carriers (electrons or holes) play an important role in stabilizing the ferromagnetism of DMS. In order to confirm above model, we also perform first-principles calculations based on the local spin density approximations to study the magnetic properties of ZnS:Fe system. In this work, the Zn$_{1-x}$Fe$_x$S thin films are prepared by low-pressure Metalorganic Chemical Vapor Deposition (MOCVD) equipment on Si substrates [7]. Structural and morphological studies are done by means of X-ray diffraction (XRD) technique using Cu Kα irradiation on an 8000 Philips 1830 powder diffractometer. X-ray Photoelectron Spectroscopy (XPS) measurement is carried out on VGESCALAB MK II instrument where Mg Ka X-ray (hν=1253.6eV) is used as the emission source. The magnetization measurements are performed by a Quantum Design superconducting quantum interference device (SQUID) system [1-6].

Experiments and Theory

ZnS:Fe single-crystal thin films are grown on Si substrates by low-pressure CBD at substrate temperature of 350 °C. The substrates are cleaned by ultrasonics with a sequence of trichloroethylene, acetone, ethanol and deionized (DI) water, and etched in the mixture solution of a H$_2$SO$_4$:$H_2$O$_2$:H$_2$O=3:1:1 at room temperature for 10 min. Before put into the growth chamber, the chemical etched substrates are rinsed in DI water again and finally dried by nitrogen blowing. Precursors consist of ironpentacarbonyl (FeCl)$_3$, dimethylzinc (DMZn) and H$_2$S. In our experiment, the growth pressure fixed is 76 Torr using a homemade...
horizontal reactor, and high purity hydrogen is used to carry the organometallics into the reaction chamber with the total gas flow rate of 21 min$^{-1}$. The Si substrate is cleaned in hydrogen ambient at 600 °C 10 min, which is demonstrated to be an effective method to remove the oxide layer and residual surface contaminants. Films are grown for 30 min and the film thicknesses are about 200 nm. The Fe content in the reactor is controlled by varying the H$_2$ flow rate. The resulting Fe concentration in the sample is determined using XPS. The structure of the ZnS:Fe thin films is characterized by X-ray diffractometers. Surface morphology of the films is investigated by AFM operated in tapping mode. The high-temperature (100 K) magnetic properties have been assessed using a SQUID magnetometer.

The calculations are performed using VASP code, based on the spin density functional theory. For the exchange and correlation potential, the generalized gradient approximation (GGA) is used. All plane waves with a cutoff energy of 300 eV are used in the basis function. For all the Fe-doped systems, a 64 atoms supercell is used for the defect calculations. The lattice constants of the supercells are kept fixed to that of pure ZnS.

Results and Discussion

The structure of ZnS:Fe thin film is examined by θ-2θ X-ray diffraction system using CuKα radiation in the range 20°-80°, which is shown in Figure 1. Firstly, strong ZnS:Fe (200) and (400) alloy peaks are observed at 2θ=33.01° and 2θ=69.30°. Si substrates (200) and (400) peak are apparent near 2θ=31.61° and 2θ=66.04°, respectively. No other peaks above the noise of the background are detected. From the Figure 1 rocking curve of ZnS (200) diffraction peak, it is clearly that the diffraction peaks indicate that ZnS single crystal grown on GaAs (200) substrate with cubic structure. And the ZnS:Fe thin film is phase pure with a (200) curve FWHM of 0.34 degrees. It can be explained that Fe ions substitute partly Zn ions and lead to the lattice structure difference. Addition, the X-ray diffraction results are comparable with those of ZnS films grown on Al$_2$O$_3$ sapphire substrates under the same conditions. It is obviously shown that ZnS single crystal grown on Al$_2$O$_3$ (006) substrate is hexagonal structure, but the sample grown on Si(200) is cubic structure. The ZnS structure relates to the substrate and the Fe content, and the lattice structure is gradually changes from hexagonal to cubic structure under different substrates.

The fraction of Fe incorporated into the ZnS singlephase film is determined using X-ray photoelectron spectroscopy (XPS). Data of the XPS indicate the relative concentration of Fe to Zn to be 11% for this sample.

AFM is used to investigate the effect of Fe doping on the ZnS single crystal thin film morphology. Single phase films are typically non-specular and exhibit large values for the root mean squared roughness, which shows in Figure 2. The morphology of the multi-phase films exhibits a much different morphology. A 2 nm×2 nm AFM micrograph for the Fe-doped ZnS thin films during the process of CBD is exhibited in Figure 2. The film thickness is 200 nm and no precipitates like Fe cluster. The temperature-dependent magnetization (M-T) curve for single-phase Fe-doped ZnS sample are measured using a SQUID magnetometer with a magnetic field of 10000 Oe, applied perpendicular to the film plane. Measurements are taken at the relatively large applied magnetic field in order to increase the magnetic signal from thin film with respect to the large diamagnetic response of the Si substrate. Measurements have also been taken at 30000 Oe and shown similar temperature dependence. The results of magnetization subtract of the diamagnetic contribution of the GaAs substrate. The dc zero-field-cooled (ZFC) and the field-cooled (FC) magnetization curves are shown in Figure 3. Measurements are performed from 5 to 300 K. The FC is obtained by measuring the magnetic moment of the sample in a magnetic field of 1000 Oe during cooling. The ZFC measurement
is obtained by first cooling the sample to 5K in zero fields and then warming it in the same field as that of the FC measurement. The ZFC magnetization shows stronger temperature dependence than the FC one below 30K. ZFC and the FC magnetization curves are measured overlap after high temperature about 180K with a paramagnetic behavior. This temperature below which the ZFC-FC curves separate, the ferromagnetic ordering sets are shown clearly in Figure 3 [2-4] displays a ferromagnetic behavior persisting up to 270K. Therefore, it obtains the Curie temperature $T_c$ of 270K in this system. As seen in Figure 3 the sample displays ferromagnetic ordering and a decrease with increasing of temperature. Moreover, ferromagnetic transition temperatures of previous report are focus in other material. For example, A transition temperature more than room temperature is obtained in Fe-doped in ZnO:Cu reported by S J [5]. In the Co-doped ZnS samples, room-temperature magnetic hysteresis is observed [2]. But up to date, there is no work reporting Fe-doped ZnS thin film with high $T_c$ of 270K grown by MOCVD. Figure 4 displays the results of magnetization as a function of applied magnetic field measured for ZnS:Fe film at 100K. The total magnetization could be described as an algebraic sum of various contributions. The substrate diamagnetic contribution is subtracted from the total magnetization signal. The remaining data consist of the paramagnetic and ferromagnetic contributions from the ZnS:Fe thin film. The well-defined hysteresis loops show that the ZnxFe1-xS film are clearly ferromagnetic at 100K. At 100K, the saturation magnetization $(MS)$ is $1.82\times10^{-5}$emu with a remanence magnetization $(MR)$ of $0.31\times10^{-5}$emu and a coercive field $(HC)$ of 50 Oe. GaAs substrate and ZnS are diamagnetic, so the magnetization data indicate that the ferromagnetization in the present study is because Fe-doped in ZnS thin film and formed ZnxFe1-xS DMS [8].
The origin of the ferromagnetism in DMS materials is still not clearly understood. To further demonstrate that the Fe-doped ZnS thin films are favorable for high temperature ferromagnetism, we model 64-atom cell to simulate the magnetic interactions of Fe atoms by performing first-principles spin-polarized Density Functional Theory (DFT) calculations within Generalized-Gradient Approximation (GGA). We can find that these magnetic interactions are mainly strong p-d coupling between the Fe 3d states and S 2p states. The calculated total energy differences between ferromagnetic (FM) and anti-ferromagnetic (AFM) configuration (ΔE=EFM-EAFM) are 46meV, indicating that ferromagnetic ordering is favorable for ZnS:Fe system. Therefore, the observed ferromagnetism should be the intrinsic behavior of Fe-doped ZnS thin film.

**Conclusion**

In summary, we have reported the ferromagnetic properties of single-phase ZnS:Fe thin films prepared on Si substrates by CBD as a new method. Field and temperature dependent magnetization curve indicated that a high Curie temperature $T_c=270K$. The XRD and AFM used for investigating the structural and surface characterization, the results indicated that no Fe clusters and other compounds are formed in the samples. The experimentally observed high $T_c$ could be mainly ascribed to the p-d exchange coupling between Fe ions and host elements.

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