Technologies for producing syntactic carbon foams with specified operational properties

Engel Galimov¹,∗, Elmira Sharafutdinova¹, and Nazira Galimova¹

¹Kazan National Research Technical University named after A.N. Tupolev–KAI, 420111, Kazan, Russia

Abstract. Technologies for producing two classes of syntactic carbon foams of regular structure were developed: thermal insulating and heat-conducting foams with specific physico-mechanical properties designed for operation under extreme conditions. The possibility of controlled development of porous structure in carbon foams upon their production by selecting the initial components in the form of binders, hollow particles (microspheres) and various additives (solvents, etc.), their optimal combination and ratio, as well as operating parameters of thermal processing. Phenol-formaldehyde resins, melamine and coal or oil pitches filled with hollow particles in the form of phenolic and carbon microspheres were used as starting components for the production of carbon foams. Technologies for producing carbon foams include the sequence of the following stages: preparation and mixing of the starting components, molding of samples, carbonization, graphitization, pyrocompaction and mechanical processing. Thermal insulation foams were obtained using two technologies. The first technology for producing foams involves using phenolic microspheres and a binder in the form of a phenol-formaldehyde resin of the novolac type. The second technology for producing foams through template carbonization involves the use of high porosity melamine with cellular structure as a binder, and hollow phenolic microspheres.

1 Introduction

Carbon foams (CF) fall into the class of modern composite materials with a high value of total porosity (60-70% and more). A specific feature of the CF is the possibility of directional development of the porous structure during the technological process by selecting the initial components in the form of fillers, binders and various additives (solvents, etc.), their optimal combination and ratio, as well as operating parameters for heat treatment (carbonization, graphitization, pyrocompaction). In the process of producing the foams, it is possible to purposefully control the main parameters of the porous structure which have a significant influence on the change in thermophysical, physical and mechanical and other properties of materials [1-5].

∗ Corresponding author: engelgalimov@yandex.ru
Among carbon foams, syntactic carbon foams (SCF) obtained using the hollow particles (microspheres) and binders in the form of polymers, pitches, and etc. as the initial components, take a special place in terms of their efficiency and perspectives of application. Such foams are characterized by the regularity of the structure, i.e., an elementary cell can be distinguished, and the structure of the whole material can be described by its multiple reiterations in space. The cells are macropores of approximately spherical shape and “windows” that provide the connection between the cells and the formation of a single open system of pores.

Research aimed at developing new technologies for producing foams with given thermophysical, mechanical and physical properties designed to operate in extreme conditions are relevant. Internationally, a number of industrial and laboratory technologies based on the carbonization of synthetic polymers filled with microspheres, foaming of carbon-containing substances followed by the carbonization, template carbonization of organic substances and polymers, compacting of expanded graphite, self-assembly of carbon nanoparticles, direct carbonization of natural raw materials, and etc. [1–4] are used for producing SCF.

The aim of the work is the development of technologies for producing heat-insulating and heat-conducting SCF with the given mechanical and physical properties. In order to achieve this goal, the following main problem was solved: to develop technology for producing two main classes of carbon foams: glass carbon-based heat-insulating carbon foams having at an apparent density of 0.1–1.0 g/cm$^3$, the ultimate compressive strength of at least 1.0 MPa and the thermal conductivity coefficient not more than 0.1–0.2 W/m·K; heat-conducting syntactic carbon foams based on graphitizing including mesophase-forming materials which have at an apparent density 0.6 g/cm$^3$, the ultimate compressive strength of at least 1.0 MPa and the thermal conductivity coefficient of not less than 150 W/m·K. It is necessary to select the initial components, determine their combinations and ratios, use specialized equipment, determine the optimal operating parameters of the technological process for producing SCF, and conduct research on their mechanical and physical, and thermophysical properties to do this.

### 1.1 Initial components for producing foams

Phenol-formaldehyde resins (PFR), melamine and pitches were used as a dispersion medium in for producing the foams. PFRs are products of phenol and formaldehyde polycondensation. Depending on the conditions of polycondensation, resol or novolac resins are formed; when they are heated the curing reaction with the formation of the polymer with a reticulate structure occurs. Melamine is a material based on hardened melamine formaldehyde resin (MFR). Pitches are remainders of the processing of coal-tar raw material or oil stock which are semi-solid amorphous substances with a softening temperature above the room temperature and characterized by a high carbon yield during heat treatment in an inert atmosphere (coking). Coal tar pitches are products of resin processing of high-temperature coking of black coal and are solids. Oil pitches are obtained from the remainders of the resin obtained in the process of oil pyrolysis. The anisotropic liquid crystal phase called the mesophase [1–3] is formed during the pyrolysis of isotropic coal tar or oil pitches in the temperature range of 300-500 °C.

Phenol formaldehyde (phenolic) and carbon microspheres were used as the dispersed phase of foams. Phenolic microspheres are produced by heat treatment of sprayed solutions or emulsions. Dilute solutions of resol PFRs containing up to 50 wt. % of water and 6-9 wt. % of phenol are used to produce spherical-shaped particles. The production process includes the solvent evaporation and the heating of the sprayed monolithic particle, the expansion of vapors or gases inside the particle and the final curing of the microsphere. Microspheres are
loose powder with the particle size of 10-300 µm and bulk density of 100-1500 kg/m³. The carbon microspheres are produced by carbonization of microspheres based on PFR at 900 °C in an inert gas environment. The microspheres have the diameter of 5-150 µm, the wall thickness of 1-4 µm and bulk density of 130-140 kg/m³ [1-3].

2 Experimental part

Heat insulating foams were produced using two technologies. The first technology for producing SCFs is based on the application of phenol microspheres using the “filler-binder” technology which includes the mixing of components, pressing of the mixture, heat treatment of the pressed workpieces, their pyrocompaction, and mechanical treatment to produce workpieces for testing. The PFR-based microspheres which are hollow spheres with a wall thickness of 1-2 µm were used as the filler. The average outer diameter of the microspheres was 34 µm. The novolac type PFRs were used as the binder. The second technology for producing the foams by templating carbonization involves the use of porous melamine of cellular structure (porosity 99.4%) from which the samples of size from 10×10×20 mm to 20×30×30 mm were made. Since the yield of carbon residue from melamine is only 7-8 wt. % then at producing the foams its use as the template, i.e., the sacrificial polymer coated by material with the high yield of carbon residue. The samples were placed in a container filled with the solution of PFR in ethanol or acetone, the container was placed in a vacuum desiccator connected to the diaphragm pump (ILMVC LVS-105T, residual pressure 8 mm Hg). The container was evacuated before the start of gas evolution, the exposure for 3-5 minutes for the gas yield was made, then the vacuum level was raised by 5 mm Hg, bringing the residual pressure below the boiling point of the PFR solvent (210 mm Hg in case of acetone and 150 mm Hg for ethanol), and then the exposure for 20 minutes was made. The PFR solutions in ethanol and acetone were prepared by heating the weighed portion of the resin in a solvent by mixing at 45°C and 65°C in the case of using, respectively, acetone and ethanol until complete dissolution with the exposure for 15 minutes at the final temperature.

Heat-conducting foams were produced using various technologies. The first technology is based on using the carbon microspheres and coal tar or oil pitch as the base components which are mixed at given ratios the solvent (toluene) is added to the mixture, heated in an oil bath, boiled using the backflow condenser at mixing after that the mixture is cooled, evaporated using the rotary evaporator to remove the toluene and dried under vacuum. The resulting mixture in the form of a granulated material is crushed and the “green” foam is molded by pressing the powder in the matrix under optimal conditions. Then, the workpiece is extracted and subjected to graphitization in a furnace at 2700 °C the graphitized workpiece is placed in a pyrocompaction furnace which is sealed, blown through with argon, the samples are heated and maintained at 800-1100 °C, the pressure of 8-12 mm Hg within 20-120 hours.

The second technology for producing the foams by carbonization of pitches under pressure is based on the use of coal tar or oil pitch with the mesophase yield of at least 20 wt. %. The mesophase microspheres produced by heat treatment of the initial pitch up to a temperature of mesophase formation (400-500°C) followed by extraction of the soluble portion in toluene are added to the mixture to increase the thermal conductivity of the foams. The mesophase microspheres are added to increase the mesophase yield from pitch up to 40-50 wt. %. The mixing of the components is carried out in a mixer at a temperature above the softening temperature of the pitch. Toluene is added to the mixture, mixed, boiled with backflow condenser for 60 minutes, and cooled, then, the solvent is extracted by distillation under vacuum using the rotary evaporator, drying is carried out under vacuum using a rotary evaporator at the temperature of 120 °C and residual pressure of 10 mm Hg for 60 minutes. After mixing the mixture is cooled and crushed in a vibrating mill, then, low-temperature carbonization of the “green” foam is conducted under pressure at 900 °C. The obtained
samples are placed in graphite crucibles with lids which are placed in an electrovacuum furnace. The working medium is pumped out to the residual pressure of less than 1 mm Hg. The furnace is heated for 8 hours up to 2100°C the samples are exposed for 2 hours then cooled and extracted. The final stages are graphitization at 2700 °C and pyrocompaction in the furnace which is sealed, blown through with argon, the samples are heated at 1100 °C and the pressure of 8-12 mm Hg within 20-120 hours.

The third technology for producing the foams by the carbonization of pitches under pressure is based on the introduction of the volatile substances into their composition at temperatures above 600 °C. The initial components are the pitches and the mesophase microspheres produced by heat treatment of the pitches up to a temperature of mesophase formation (400-500°C), the sodium chloride is used as the pore agent. The component mixing is carried out by joint grinding in a ball mill for 7 minutes. Then the molding powder is placed in a heated matrix and molded under pressure for 3 minutes. The resulting workpiece is extracted and carbonized. Next, samples of “green” foam are placed in a steel container, covered by the layer of broken graphite, a layer of tar pitch is poured on top, this is covered with a lid, placed in the furnace and heated at the rate of 2.5°C/min up to 900°C for 120 minutes, after that the furnace is cooled. The resulting samples are placed in graphite crucibles with lids, and filling is poured. The crucibles are placed in an electrovacuum furnace the working medium is pumped out to the pressure of less than 1 mm Hg. The furnace is heated for 8 hours up to 2100°C the samples are exposed for 2 hours then cooled. For graphitization, the samples are placed in graphite crucibles with lids, which are placed in a graphitization furnace with the temperature of 2700°C and the samples are exposed for 60 minutes. Then, the samples are loaded into a pyrocompaction furnace which is sealed, blown through with argon, heated and exposed at 1100 °C and the pressure of 8-12 mm Hg within 20-120 hours. For the obtained samples, the apparent density, porosity, compressive strength, and thermal conductivity coefficient were determined using standard techniques [5-7].

The table 1 shows the comparison of the obtained experimental results (variants 1-5) with theoretical and published data (variants 6-12).

| No. | Variants of technology | Density, g/cm³ | Porosity, % | Strength, MPa | thermal conductivity coefficient, W/m·K |
|-----|------------------------|----------------|-------------|---------------|--------------------------------------|
| 1   | Foams produced on the basis of the carbon microspheres and PFR | 0.07-0.80 | 64-99 | 5.2-23 | 0.015-2.6 |
| 2   | Foam produced by template carbonization | 0.16-0.63 | 71-93 | 5.2-23 | 0.9-3.1 |
| 3   | Foams produced on the basis of the carbon microspheres and pitches | 0.25-0.67 | 69.5-89 | 9.6-19.2 | 10-50 |
| 4   | Foams produced by carbonization under pressure | 0.88-1.08 | 51-60 | 26-33 | 139.5-162.0 |
| 5   | Foams produced using the volatile pore agent | 0.21-0.39 | 82-91 | 5.4-12.6 | 108-172 |
| 6   | Foams produced on the basis of carbon fiber P55 | 0.55 | 75 | 15 | 12 |
| 7   | Foams produced on the basis of carbon fiber P120 | 0.55 | 75 | 25 | 60 |
| 8   | Carbon foam ORNL | 0.60 | 73 | 20-25 | 180 |
| 9   | Foam produced of pitch of brand AR (2800°C) | 0.56 | 75 | 30 | 187 |
| 10  | Graphtized foam Conoco (2800°C) | 0.59 | 73 | ~30 | 134 |
| 11  | Copper foam | 2.25 | 75 | 68 | 40 |
| 12  | Aluminum foam 6061 | 0.50 | 81 | 28 | 15 |
From the presented data it is clear that the foams produced using the technologies proposed in the paper surpass the majority of described analogues both on the basis of carbon and on the basis of polymers and metals per totality of the studied properties.

Based on the analysis of the conducted research, the possibility to use an integrated technology for producing heat-insulating foams with the thermal conductivity coefficient of less than 0.1–10 W/m·K (SCF-1), highly heat-conductive foams with thermal conductivity coefficient from 50 to 160 W/m·K (SCF-3), as well as foams with the intermediate value of thermal conductivity coefficient from 10 to 50 W/m·K (SCF-2) was established. The research results showed that the technology using carbon or phenolic microspheres and PFRs as a binder could be applied for the production of SCF-1. For the production of SCF-2, the technology based on pitches carbonization under pressure is the most optimal. Despite the significant difference of the early conversions (up to low-temperature carbonization) the same furnace equipment as for the previous approaches can be used for this one, which makes it possible to organize a flexible technology for the production of foams with various properties. Moreover, the same method of mixing (in a solvent, in a reactor with a stirrer, followed by evaporation using a rotary evaporator) is used for the production of foams with low and medium thermal conductivity coefficient. The processes similarity of subsequent heat treatment of the compositions provides the technical and economic advantages of the integrated technology as it allows significant reduction of the amount of the used processing equipment. The following equipment is necessary to organize a unified technological scheme for producing foams with various properties: a reactor with a stirrer and backflow condenser (SCF-1, SCF-2); mixing machine with Z-shaped blades (SCF-3); low-temperature furnace/drying furnace with heating up to the temperature of at least 150°C (SCF-1); ball mill (SCF-1 – SCF-3); hydraulic press (SCF-1, SCF-2); high-temperature autoclave (550°C; 15 MPa) (SCF-3); kiln (up to the temperature of 1000°C) (SCF-1 – SCF-3); electrovacuum furnace (up to the temperature of 2200°C) (SCF-1 – SCF-3); graphitization furnace (up to the temperature of 3000°C) (SCF-2, SCF-3); pyrocompaction reactor (SCF-1 – SCF-3).

The figure 1 shows the scheme for producing foams using integrated technology.

![Scheme of integrated technology for producing foams](image)

**Fig. 1. Scheme of integrated technology for producing foams**

### 2.1 Practical significance

SCFs can be used to control heat flows, create prospective current sources, as well as producing articles and elements of constructions in the defense technologies, aviation, and space; appropriateness and relevance of the technology for producing articles with given thermophysical properties for operating in extreme conditions based on the synthetic resins...
are shown; recommendations on the use of the research results in industrial enterprises are given.

3 Conclusions

Raw materials that provide a high degree of graphitization during heat treatment should be used to produce foams with high values of thermal conductivity coefficient. The necessary condition for this is the formation of a viscoplastic phase during carbonization, an example of which is the mesophase as a liquid-crystalline state of a substance. This fact necessitates the use of mesophase-forming pitches (coal tar and oil) to produce articles with high values of thermal conductivity coefficient. In addition, it is possible to use additives of mesophase particles (microspheres) which can increase the average degree of material graphitization to increase the mechanical and physical properties and thermal conductivity coefficient of SCF. The regulation of the SCF macrostructure based on the graphitized materials, i.e., the creation of a regular system of spherical pores is possible through the use of additives of phenolic or carbon microspheres.

Synthetic polymeric binders (PFR, melamine formaldehyde resins) during carbonization form carbon characterized by the low thermal conductivity coefficient (not more than 10 W/m·K), so they are suitable for producing the heat-insulating SCFs. At the same time, the cellular polymers of honeycomb structure for producing the SCFs by the template synthesis method (melamine) are characterized by the low yield of carbon residue, which can lead to low mechanical and physical properties of the obtained articles. On the other hand, PFRs characterized by the high yields of carbon residue (up to 95 wt. %) require the use of structuring additives and fillers (microspheres) to create materials with regular porous structure.

A unified integrated production technology that allows producing a wide range of ultra-high-temperature carbon foams, including those with a density of less than 0.6 g/cm³ with the possibility of directional control of the thermal conductivity coefficient in the range from less than 0.1 W/m·K to 160 W/m·K, was developed.

It was established that the processes parameters of producing the foams have a significant impact on the change in their properties and can be used as an effective factor in the directional control of the properties of the studied class of materials. From a comparison with the literature data, it is obvious that by the basic properties the obtained foams are on the level or significantly exceed the world analogs.

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