Effective photocatalysis of functional nanocomposites based on carbon and TiO$_2$ nanoparticles†

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A unique nanocomposite C–TiO$_2$ was prepared by the growth of TiO$_2$ on carbon nanoparticles using a simple hydrothermal procedure. Transmission electron microscopic (TEM) measurements showed that the nanocomposites exhibited an average core diameter of approximately 5 nm with a rather well-defined lattice space (0.4 nm) that was somewhat larger than that (0.38 nm) of the (100) crystalline planes of anatase TiO$_2$. This lattice expansion was accounted for by the formation of surface defect dipoles of the nanosized TiO$_2$ particles. X-ray photoelectron spectroscopic (XPS) measurements suggested that partial charge transfer occurred from carbon nanoparticles to TiO$_2$ by the interfacial Ti–O–C linkages, which led to effective diminishment of the C–TiO$_2$ photoluminescence as compared to that of pure TiO$_2$ or carbon nanoparticles, suggesting intimate electronic interactions between the carbon and TiO$_2$ components in the nanocomposites. Such unique characteristics were then exploited for the effective photocatalytic degradation of organic pollutants, as exemplified by methylene blue, by C–TiO$_2$ under UV photoirradiation. Experimental measurements showed that the photocatalytic activity of C–TiO$_2$ nanocomposites was about twice that of TiO$_2$ alone, whereas little activity was observed with carbon nanoparticles. This was attributed to the electron-accepting sites on the carbon nanoparticles that facilitated interfacial charge separation.

Introduction

Titanium dioxide (TiO$_2$) has been extensively studied for potential applications in photocatalysis due to its low cost, non-toxicity, high redox ability and stability.$^1$–$^4$ However, with an intrinsic bandgap close to $\sim$3 eV, TiO$_2$ typically exhibits low efficiency in light utilization and fast recombination of electron–hole pairs, which greatly limit the photocatalytic performance. Therefore, in the past few decades, a number of effective approaches have been developed to further engineer the TiO$_2$ structures and properties, such as coupling with other semiconductor materials,$^5$–$^7$ modification with metal and nonmetal elements and structures,$^8$–$^{12}$ and surface sensitization by dye molecules.$^{13}$–$^{15}$ Among these, formation of functional composites with carbon nanomaterials represents one of the most popularly used methods, largely motivated by the unique optical and electronic properties of carbon nanostructures. There are various kinds of carbon modifications, such as carbon-doped TiO$_2$,$^{16}$–$^{20}$ carbon-coated TiO$_2$,$^{21}$–$^{24}$ and mounting of TiO$_2$ on activated carbon.$^{25}$–$^{27}$ For instance, Chu et al.$^{17}$ prepared carbon-doped nanostructured TiO$_2$ films (the average size of the nanoparticles was about 40 nm) with oxalic acid and tetrabuty-lammonium bromide as the carbon sources and the carbon-doped into the lattice sites of TiO$_2$, leading to apparent narrowing of the TiO$_2$ bandgap and hence improved efficiency of photoenergy utilization. In another study,$^{28}$ carbon-doped TiO$_2$ nanotubes, nanowires and nanorods were prepared by utilizing the nanoconfinement of hollow titanate nanotubes. The procedure entailed two steps, adsorption of ethanol onto the inner walls of the nanotubes and thermal treatment of the complex in an inert nitrogen atmosphere. The resulting catalysts exhibited enhanced photocatalytic activity in the degradation of toluene in the gas phase under both visible and simulated solar light irradiation, as compared to that of commercial Degussa P25. Carbon dopants have also been incorporated into TiO$_2$ micro/nanospheres and nanotubes by chemical vapor deposition in an inert atmosphere, and the resulting TiO$_2$ exhibited much higher photocurrents than commercial P25.$^{28,29}$ In these studies, the key motivation is to engineer the TiO$_2$ bandgap by carbon dopants so that the efficiency of light absorption and utilization can be improved. However, the synthetic procedure typically requires rather harsh experimental conditions and/or sophisticated instrumentation, and the lack of structural stability of the dopants might compromise the durability of the photocatalysts.

Another effective strategy is to prepare carbon and TiO$_2$ nanocomposites. For instance, carbon-coated anatase TiO$_2$
catalysts (>20 nm in diameter) have been prepared by heating a mixture of poly(vinyl alcohol) and commercial TiO$_2$ at elevated temperatures (700–900 °C), which exhibited markedly enhanced photocatalytic activity in the degradation of methylene blue, as compared to that of the original commercial TiO$_2$.

**Synthesis of TiO$_2$ nanoparticles**

TiO$_2$ nanoparticles were prepared by a hydrothermal method. Briefly, 0.1 mL of 3-chloroaniline was added into 15 mL of Nanopure water, and then transferred to a Teflon-lined stainless-steel autoclave. Subsequently, 0.17 g of titanium(IV) $n$-butoxide and 1.0 mL of oleic acid were dissolved in 10 mL of toluene and then transferred to the autoclave. The mixed solution was then subject to hydrothermal reactions at 180 °C for 12 h. The white precipitates were collected and extensively rinsed with methanol.

**Synthesis of carbon hybridized TiO$_2$ nanoparticles (C–TiO$_2$)**

The preparation and processing of carbon hybridized TiO$_2$ was the same as that of TiO$_2$ described above. The only difference was to add 15 mg of carbon nanoparticles to the water phase at a concentration of 1 mg mL$^{-1}$ prior to the hydrothermal procedure. The resulting C–TiO$_2$ nanocomposite contained about 27 wt% of carbon.

**Characterization**

Transmission electron microscopy (TEM) studies were carried out with a JOEL JEM-2010 TEM microscope operated at 200 kV. The samples were prepared by suspending the powder sample in toluene and dropcasting onto a Cu grid. UV-vis absorption spectra of the samples were measured with an SCI Unicam UV4 spectrometer by using a 1 cm quartz cuvette with a resolution of 2 nm. Fluorescence spectra were acquired with a PTI photoluminescence spectrometer. X-ray photoelectron spectra (XPS) were recorded with a PHI 5400 XPS instrument equipped with an Al K$_\alpha$ source operated at 350 W and at 10$^{-5}$ Torr. The spectra were charge-referenced to the Au4f$_{7/2}$ peak (83.8 eV) of sputtered gold.

**Photocatalysis**

Methylene blue at the concentration of 20 mg L$^{-1}$ was used as the testing solution, and a UV lamp (365 nm, 6 W) was employed as the light source. In a typical experiment, 2 mL of the methylene blue solution was added into a quartz cuvette and 1 mg of photocatalysts was added into the solution and mixed under magnetic stirring for at least 2 h in the dark to establish an adsorption/desorption equilibrium, prior to exposure to the light source. In photocatalytic assessments, the distance between the light source and the cuvette was ca. 10 cm. The solution was under magnetic stirring during the entire experimental procedure. The photocatalytic activities were quantified by collecting the UV-vis absorption spectra of the solution at different reaction intervals.

**Results and discussion**

Fig. 1 depicts two representative TEM micrographs of (A and B) TiO$_2$ and (C and D) C–TiO$_2$ nanoparticles. Despite rather
extensive agglomeration in both samples (panels A and C), individual particles remained resolvable with a largely spherical shape and an average diameter of 5.20 nm and 5.05 nm for the TiO2 and C-TiO2 nanoparticles, respectively. In addition, high-resolution TEM measurements exhibited well-defined lattice fringes in both TiO2 and C-TiO2 nanoparticles, where the lattice spacing was estimated to be ca. 0.40 nm (red lines). This is somewhat larger than the d-spacing (0.38 nm) of the (100) diffracting planes of bulk anatase TiO2.34 In a previous study it was found that TiO2 exhibited an apparent linear lattice expansion with a diminishment of the grain size to the nanometer regime (<50 nm), as a result of the formation of surface defect dipoles, in contrast with metal nanoparticles that typically exhibited positive surface pressure and hence lattice contraction instead with shrinking physical dimensions.35 In the present study, the TiO2 nanoparticles were very small, only about 5 nm in diameter. Because of low coordination of the surface sites, bonding interactions with reactive molecules (e.g., H2O) likely occurred, leading to the generation of surface defect dipoles and lattice expansion, as observed above. Nevertheless, one can see that in both samples the TiO2 nanoparticles were of rather pure crystalline phase and rich in high-energy {100} facets, consistent with X-ray diffraction studies (Fig. S1†). This may be advantageous to their photocatalytic performance in the degradation of organic pollutants, as detailed below.34

In the above TEM study, it was difficult to resolve the carbon components in C-TiO2 probably because of the low contrast and poor crystallinity as compared to TiO2.32 Yet the electronic interactions between the carbon and TiO2 components in the C-TiO2 hybrids were readily revealed by XPS measurements. Fig. 2(A) depicts the full survey spectra of the TiO2 and C-TiO2 nanoparticles. One can see that in both samples the elements of C, Ti, and O can be clearly identified, with the binding energies of C1s, Ti2p and O1s electrons at approximately 286.6 eV, 459.4 eV and 530.7 eV, respectively (the carbon signals in TiO2 were likely from the organic precursors used to prepare the nanoparticles, whereas in C-TiO2, there were additional contributions from the carbon nanoparticles). However, high-resolution scans showed that the exact binding energies of these electrons were slightly different, as revealed in panels (B) to (D). In panel (B), the Ti2p1/2 and 2p3/2 electrons can be identified at 465.1 eV and 459.4 eV for the TiO2 nanoparticles (bottom curves), and slightly lower for C-TiO2 at 465.0 eV and 459.2 eV (upper curves), suggesting a partial charge transfer from functional moieties on the carbon nanoparticle surface to the Ti(IV) centers in the oxide nanoparticles. Nevertheless, these binding energies (and their corresponding spin–orbital splitting) are highly comparable to those reported in the literature for TiO2. For instance, in a previous study, for anatase TiO2 grown on multiwalled carbon nanotubes (MWCNT) the binding

![Representative TEM micrographs of (A and B) TiO2 and (C and D) C-TiO2 nanoparticles. Red lines highlight the lattice fringes of TiO2.](image)
energies for Ti2p electrons were found at 464.9 eV and 458.8 eV, whereas for bulk anatase TiO2 at 464.5 eV and 458.8 eV, respectively.

Similar behaviors were observed with the binding energies of the C1s electrons. From panel (C), one can see that for TiO2 nanoparticles (bottom curves), there are two major peaks centered at 285.4 eV and 286.8 eV, which may be assigned to the 1s electrons of sp² hybridized carbon (C–C, for instance, from the aniline precursors used in TiO2 nanoparticle synthesis) and to the formation of Ti–O–C carbonate species, respectively. For the C–TiO2 nanoparticles (top curves), these two peaks red-shifted slightly to 285.0 eV and 286.7 eV. The significantly greater intensity of the higher energy peak can be ascribed to the presence of carbon nanoparticles where the surface was decorated with various oxygenated functional species and they served as the anchoring sites for TiO2 nanoparticle growth. Furthermore, a small broad peak can be seen at around 290 eV, which may be assigned to the carboxylate C1s electrons on the carbon nanoparticle surface.

For the O1s electrons, one can see from panel (D) that both TiO2 and C–TiO2 nanoparticles exhibited an asymmetric peak within the range of 528 eV to 538 eV, whereas for bulk anatase TiO2 at 533.3 eV, which most probably included contributions from both the TiO2 hydroxyl groups as well as various oxygenated species on the carbon nanoparticle surfaces (such as carbonyl, carboxylate, phenylate, etc.). This again indicates that carbon nanoparticles were indeed incorporated with TiO2 to form nanoscale composites.

The intimate electronic interactions between the carbon and TiO2 nanoparticles in C–TiO2 led to unique interfacial charge transfer upon photoirradiation. Fig. 3 shows the excitation (solid curves) and emission (dashed curves) fluorescence spectra of the carbon, TiO2 and C–TiO2 nanoparticles. It can be seen that the carbon nanoparticles (black curves) exhibited a broad excitation spectrum centered at ca. 311 nm and an equally broad emission peak at approximately 428 nm, a behavior consistent with the results reported previously that was ascribed to electronic transitions of oxygenated functional species on the nanoparticle surface.

For TiO2 nanoparticles (red curves), the fluorescence features were different, where a major excitation peak might be identified at 362 nm, along with two emission peaks at 460 nm and 528 nm. Note that the excitation peak energy (3.4 eV) is somewhat greater than the bandgap energy of bulk anatase TiO2 (3.2 eV), most probably as a result of lattice expansion as observed above (Fig. 1). It should be mentioned that reports of photoluminescence emission of TiO2 nanoparticles are generally rather few, as titania is a well-known indirect bandgap semiconductor and in most previous studies the nanoparticles possess low crystallinity and broad size distributions. In an earlier study, using a sol–gel method we prepared TiO2 nanoparticles (5 nm in diameter) that were attached onto a gold Janus nanoparticle.
surface and observed apparent fluorescence characteristics with the excitation and emission peaks at ca. 400 and 609 nm, respectively, which were ascribed to electronic transitions involving trap states of the nanoparticles. However, for the C–TiO$_2$ nanoparticles (green curves), the fluorescence intensity was drastically diminished in both excitation and emission measurements, with the excitation and emission peaks at 360 nm and 403 nm, respectively. Note that the former is very close to that of TiO$_2$ nanoparticles whereas the latter appears to be part of the broad emission envelope of carbon nanoparticles. This suggests that in the composite C–TiO$_2$ nanoparticles, efficient charge/energy transfer likely occurred from TiO$_2$ to carbon under photoirradiation, leading to apparent quenching of the TiO$_2$ fluorescence. This may be facilitated by the oxygenated functional moieties on the carbon nanoparticle surface that behaved as efficient electron acceptors and were suspected to be the active sites for photoluminescence. 

In addition, as manifested in the UV-vis absorption spectra (Fig. 3, inset), one can see that whereas TiO$_2$ nanoparticles exhibited an absorption edge around 360 nm (red curve), the C–TiO$_2$ composite nanoparticles extended the absorption to the visible range (up to 450 nm, green curve), very similar to that of the carbon nanoparticles (black curve). This again indicates that carbon nanoparticles and TiO$_2$ nanoparticles were indeed integrated to form C–TiO$_2$ nanocomposites.

The photocatalytic activity of the C–TiO$_2$ nanocomposites was then examined by using the photodegradation of methylene blue as the illustrating example. Fig. 4 shows the UV-vis absorption spectra of methylene blue in water as a function of reaction times catalyzed by C–TiO$_2$ nanoparticles. The concentrations of the C–TiO$_2$ nanoparticles and MB were 0.5 mg mL$^{-1}$ and 20 mg L$^{-1}$ respectively. It can be seen that for carbon nanoparticles alone (black circles), there was virtually no change of the optical absorption even after four hours of UV photoirradiation of the methylene blue solution, indicating little photocatalytic activity. For TiO$_2$ nanoparticles (red triangles), the fraction of methylene blue that was photodegraded increased almost linearly with reaction time, and after 120 min of UV light irradiation, about 70% of photocatalytic activity of the C–TiO$_2$ nanocomposites in the degradation of methylene blue. From the optical absorbance at 664 nm ($A_{664}$), the fraction of methylene blue that was degraded could be estimated, $(A_{664} - A_{o,664})/A_{o,664}$, where $A_{o,664}$ denotes the absorbance prior to UV light irradiation (i.e., $t = 0$ min). The corresponding reaction dynamics is depicted in Fig. 5 (yellow diamonds). The photocatalytic activities of the carbon, TiO$_2$ nanoparticles and their simple mixture were also evaluated in a similar fashion and included for comparison.

![Fig. 3](image3.png) Excitation (solid curves) and emission (dashed curves) fluorescence spectra of carbon, TiO$_2$ and C–TiO$_2$ nanoparticles. The inset shows the corresponding UV-vis spectra of the three samples with the absorbance normalized to the respective value at 290 nm.

![Fig. 4](image4.png) UV-vis absorption spectra of methylene blue (MB) in water as a function of reaction times catalyzed by C–TiO$_2$ nanoparticles. The concentrations of the C–TiO$_2$ nanoparticles and MB were 0.5 mg mL$^{-1}$ and 20 mg L$^{-1}$ respectively.

![Fig. 5](image5.png) Photodegradation efficiency of carbon, TiO$_2$, and C–TiO$_2$ nanoparticles, as well as the simple mixture of carbon nanoparticles and TiO$_2$ nanoparticles (C + TiO$_2$) under UV light irradiation. Data were acquired from UV-vis absorption measurements as exemplified in Fig. 4.
methylene blue was consumed. Note that when carbon and TiO$_2$ nanoparticles were simply mixed together (with a quantity and mass ratio similar to those of C–TiO$_2$) into the reaction solution (green squares), the reactivity in the early stage of the reactions (~60 min) was almost identical to that of TiO$_2$ nanoparticles, suggesting that the TiO$_2$ nanoparticles were the major contributors in the mixture to the photocatalytic activity; at longer reaction times (>60 min), the activity of the carbon and TiO$_2$ nanoparticle mixtures actually diminished. In sharp contrast, the activity of the C–TiO$_2$ nanocomposites (yellow diamonds) was markedly better. After only 60 min of UV light photoirradiation, more than 75% of methylene blue was degraded already. That is, within the present experimental context, the photocatalytic activity of C–TiO$_2$ nanocomposites was about twice that of TiO$_2$ alone, and the performance deterioration during longer reaction times (>60 min) would be mainly attributed to the enhanced dispersion and/or crystallinity (by thermal treatments) of TiO$_2$ nanoparticles < C–TiO$_2$ nanocomposites. This indicates that intimate contacts and hence electronic interactions between the carbon and TiO$_2$ nanoparticles in the C–TiO$_2$ nanocomposites were likely to be the key factors that facilitated interfacial charge separation upon photoirradiation and hence led to improved photocatalytic activity, consistent with the effective diminishment of the photoluminescence emission of C–TiO$_2$ as compared to that of TiO$_2$ nanoparticles alone (Fig. 3).

As mentioned earlier, the carbon nanoparticle surface was decorated with various oxygenated functional moieties, including quinone derivatives, such as 9,10-phenanthraquinone, that are well-known electron acceptors. For instance, previously we observed that the carbon nanoparticles exhibited two pairs of voltammetric peaks with the formal potentials at −0.28 V vs. Ag/AgCl in electrochemical measurements. Note that in neutral solutions, the valence band of TiO$_2$ is located at ca. −0.35 V vs. Ag/AgCl whereas the valence band is at ca. +2.65 V. Thus, under UV photoirradiation, the valence electrons of TiO$_2$ were excited to the conduction band, which then crossed the carbon–TiO$_2$ interface and were accepted by the quinone moieties of the carbon nanoparticles. Such photogenerated electrons were then used to reductively degrade methylene blue, hence leading to the diminishment of the optical absorption (the corresponding holes were likely involved in the oxidation of water). For the physical mixture of carbon and TiO$_2$ nanoparticles, the lack of bonding interactions greatly impeded charge/energy transfer between them and thus they behaved independently, as observed above.

It should be noted that in earlier studies of carbon-supported TiO$_2$, the enhanced photocatalytic activities were largely attributed to the enhanced dispersion and/or crystallinity (by thermal treatments) of TiO$_2$, whereas the impacts of the chemical reactivity of the carbon support were largely ignored. In the present study, using equally nanosized TiO$_2$ particles as the comparative example, one can see that the rich surface chemistry of carbon nanoparticles might also be exploited to facilitate charge separation upon UV photoirradiation, leading to further enhancement of the photocatalytic performance.

Conclusions

A functional C–TiO$_2$ nanocomposite was prepared by the growth of TiO$_2$ on carbon nanoparticles by a simple hydrothermal procedure. XPS measurements showed that partial charge transfer occurred from carbon nanoparticles to the Ti(IV) centers in the oxide nanoparticles by the Ti–O–C interfacial bonds, in comparison to pure TiO$_2$ nanoparticles. Because of these intimate electronic interactions, the C–TiO$_2$ nanoparticles exhibited apparent diminishment of the photoluminescence as compared to TiO$_2$ and carbon nanoparticles, suggesting that effective electron/energy transfer occurred in the nanocomposites under photoirradiation. Such unique properties were exploited for the enhancement of the photocatalytic activity of C–TiO$_2$ nanocomposites, as illustrated in the photodegradation of methylene blue, which was ascribed to the improved efficiency of interfacial charge separation from TiO$_2$ to carbon because of abundant electron-accepting sites on the carbon nanoparticles.

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