I. Introduction

Following extensive research into ABO$_3$-type transition metal oxide perovskites in the 1950s, among the series LaTO$_3$ (T = Cr, Mn, Fe, Co, Ni), LaCoO$_3$ was demonstrated to exhibit rich magnetic and structural properties. For example, LaCoO$_3$ does not possess the expected high-symmetry cubic structure found in most perovskites, but has a pseudo-cubic distortion best described with the rhombohedral space group $R3c$ (no. 167) up to temperatures of ~650 K, as illustrated in Fig. 1. In addition, the magnetic susceptibility $\chi(T)$ of LaCoO$_3$ shows both thermal activation behavior and Curie-Weiss paramagnetic behavior arising presumably from the excitation of nonmagnetic Co$^{3+}$ ions ([Ar]3d$^4$) sitting in the CoO$_6$ octahedral CEF, which has been attributed to a thermally activated two-level spin state transition of Co$^{3+}$ from the low-spin (LS = t$^6_{2g}e^0_g$) state with $S = 0$ to the high-spin (HS = t$^4_{2g}e^2_g$) state with $S = 2$, although orbital ordering for the intermediate-spin (IS = t$^2_{2g}e^4_g$) state with $S = 1$ has also been proposed.$^{5,7}$ Many conflicting reports about the spin state of LaCoO$_3$ have been made based on the DFT calculations, susceptibility model fittings, and spectroscopic studies since.$^{4,8-13}$

In their pursuit of a deeper understanding of the “transitional” nature of d-orbital electrons from a localized to an itinerant character-in the transition metal perovskites, Goodenough and co-workers$^{16-19}$ have raised many fundamentally important questions closely related to LaCoO$_3$, such as (1) why is it difficult to obtain phases having more than 50% of Co$^{4+}$ through A-site substitution? (2) What is the origin of the localized spins showing “Curie tail” behavior below ~30 K? (3) Does disproportionation of Co$^{3+}$ into Co$^{2+}$–Co$^{4+}$ occur? (4) Is there an ordered configuration of LS and HS? In addition, does the IS state really exist in LaCoO$_3$? To answer these questions conclusively, mixed valence of cobalt due to nonstoichiometry must first be ruled out, especially the oxygen vacancy.

By focusing on reducing the localized impurity spin contribution in the growth of single-crystal LaCoO$_3$, we have obtained vacancy-free single crystals using the optical traveling solvent floating-zone method following a high-oxygen-pressure and high-temperature post-annealing procedure. In particular, the vacancy-free sample shows no low-temperature $\chi(T)$ Curie tail, but saturates to a constant at $T \to 0$ K, which is in sharp contrast to the behavior resulting from cobalt and/or oxygen vacancy defects that has been reported in nearly all early publications.$^{7,20,21}$ Integrated data analysis of the magnetic susceptibility $\chi(T)$, specific heat $C_p(T)$, and thermal expansion coefficient $\alpha(T)$ without complications arising from impurity...
spins and mixed valence leads to the conclusive identification of an HS population distribution across the two energy gaps of $\Delta E \sim 0.5$ meV and $Q \sim 25$ meV before and after phonon softening.

Current findings on defect-free LaCoO$_3$ are able to resolve most of the controversies raised in the literature, including the debate about the existence of an IS state and orbital ordering. In addition, the finding of three times increase of thermal expansion coefficient above the thermally activated phonon softening range strongly suggests that the nontrivial spin–orbit–phonon coupling is responsible for the pseudo-cubic symmetry of the ABO$_3$-type perovskite due to constant Co–O bond length fluctuation at room temperature.

II. Experimental details

Single crystal LaCoO$_3$ was grown using optical floating-zone growth method through direct melting and re-crystallization for its congruent melt nature with a melting point of 1750 °C. In the preparation of polycrystalline LaCoO$_3$ feed rod, Co and O loss is unavoidable during high temperature treatment in the air at ambient pressure, which is evidenced by the commonly found Al$_2$CoO$_4$ precipitate in blue color at the bottom of alumina annealing tray (shown in Fig. 2). It is found that a 5% Co excess in the precursor is required to compensate for the loss of Co for treatment above 990 °C. For the high pressure oxygen annealing process, the as-grown single crystal was mixed with Co$_3$O$_4$ powder in molar ratio of 1 : 1 sealed in a thick wall 310S stainless steel tubing with added KClO$_4$ oxidant. The sealed tubing was annealed at 1200 °C for one week. The quantitative chemical analysis has been applied to the as-prepared and annealed crystal samples with electron probe micro-analysis (EPMA) and the oxygen/nitrogen combustion analyzer (EMGA-920, Horiba).

The magnetic susceptibility, specific heat, and thermal expansion coefficient of single crystal LaCoO$_3$–$\delta$ samples were measured using SQUID-VSM (QD-USA) in both AC and DC modes, Physical Property Measurement System (PPMS, QD-USA) with a pulsed heating method, and a capacitance-type device in the PPMS platform between 2 to 600 K, respectively. In particular, the linear thermal expansion coefficient $\alpha = (1/L_0) dL/dT$ of single-crystal samples was measured directly with a newly designed capacitance-type device incorporated into the PPMS.

The electron energy-loss spectroscopy (EELS) spectra were acquired using Transmission Electronic Microscope (TEM) with a JEOL 2100F field emission gun operated at 200 kV, which equipped with a Gatan image filter (GIF, Tridiem 863) of energy resolution $\sim0.85$ eV. The TEM specimen was prepared with
focused ion beam (FIB, Tescan GAIA). The single scattering STEM-EELS spectra were deconvoluted from the raw data with subsequent Kramers–Kronig Analysis (KKA), and the calculation of effective electron number ($n_{	ext{eff}}$) was conducted with the Digital Micrograph EELS package which was written on the basis of ref. 22.

III. Results and discussions

A. Crystal nonstoichiometry

The chemical analysis results of the as-prepared and annealed crystal samples are summarized in Table 1. Both the Co and O vacancies have been controlled through the initial 5% Co excess adjustment and the post high-pressure high-temperature annealing process. The magnetic susceptibilities $\chi(T)$ and the isothermal magnetization $M(H)$ at 2 K are compared in Fig. 3 to show the impact of nonstoichiometry on the physical properties. It is noted that samples with either Co or O vacancies shows clear Curie-tail behavior for $\chi(T)$ at low temperature, which indicates the existence of localized spins from defects. Additional field scan of isothermal magnetization $M(H)$ at 2 K [inset of Fig. 3(a)] has provided solid evidence that the impurity spin of Curie-tail behavior is consistent to the paramagnetic behavior of $S = 1/2$ from the isolated spins following the Brillouin function. The total number of spin per mole is estimated from the intercept of $M(2K, H \to 0)$ for LaCoO$_{2.92}$ to be $N \sim 3.4\%$, which is consistent to the vacancy level of $\delta \sim 0.08$ within error, i.e., each oxygen vacancy roughly generates an additional spin 1/2. In contrast, the vacancy-free sample shows no Curie-tail in $\chi(T)$ and its isothermal magnetization at 2 K shows the constant paramagnetic background without contribution from the additional $S = 1/2$ spins being aligned by the increasing field.

Yan et al.,$^{23}$ proposed that the low temperature upturn could be coming from the ferromagnetic (FM) behavior of the HS state in their study. However, our results have ruled out the FM interpretation in several aspects: (1) the upturn behavior is removed entirely after high pressure oxygen annealing, which supports the explanation of oxygen vacancy-induced PM behavior, (2) no FM phase transition is identified below $\sim 20$ K from $C_p(T)$ data to support the existence of a FM phase transition, (3) the size of LaCoO$_3$ single crystal grown by the FZ method is large enough to obtain a crack-free single crystal sample for the susceptibility measurement by slicing, which significantly reduced the surface-to-bulk ratio. Thus, the surface contribution can be neglected entirely in this study. On the other hand, Pimental et al.,$^{24}$ pointed out that higher concentration point defects would prevent the valence number of Co ions to reach the perfect +3 valence. Although they proposed that antiferromagnetic coupling between Co$^{3+}$/Co$^{4+}$ may induce a weak canted-spin ferromagnetic moment, which lacks microscopic model, it’s challenging to apply their interpretation to explain our observations based on current experimental results.

Besides the magnetic property difference shown in Fig. 3(a), an obvious difference between the as-prepared and annealed samples is revealed on the transport behavior also. The onset of metal-to-insulator transition is shown shifted from $\sim 175$ K to $\sim 275$ K after the post-annealing, and the Arrhenius plot shown in the inset of Fig. 3(b) suggests that besides the intrinsic gap of $\sim 400$ meV for both, the sample with oxygen vacancies has an additional gap opening of $\sim 153$ meV, which is probably related to the formation of a defect-generated impurity band.

B. Direct measurement of $\chi^{VV}$ and spin gap $Q$

Both homogeneous DC and AC spin susceptibilities of pulverized LaCoO$_3$ ($\delta \sim 0$) crystal samples without Co/O vacancies were measured, and Fig. 4(a) shows $\chi(T)$ data using an AC field of 5 Oe/10 Hz from 2 to 500 K after the core diamagnetic correction ($\chi^{\text{core}} = \chi_{\text{La}} + \chi_{\text{Co}} + 3 \chi_{O} = -0.66 \times 10^{-4}$ cm$^3$ mol$^{-1}$)$^{15}$ to

$$\chi(T) = \chi^{\text{imp}}(T) + \chi^{\text{core}} + \chi^{VV} + \chi^{S}(T)$$

where $\chi^{\text{imp}}(T)$ is the impurity spin contribution, $\chi^{VV}$ is the Van Vleck paramagnetic contribution, and $\chi^{S}(T)$ is the actual spin contribution. Comparing susceptibilities for the high-oxygen-pressure annealed sample LaCoO$_3$ ($\delta \to 0$) with those for the as-grown sample of LaCoO$_{3.4}$ ($\delta \sim 0.08$–0.22) [see Fig. 3(a)],

**Table 1** EPMA results for LaCoO$_3$ crystals prepared under various conditions, from the stoichiometric ratio, with added 5% excess of Co, and with additional post high-oxygen-pressure annealing. The formula unit (f. u.) is converted from the averaged raw data normalized to La.

|        | La (at%) | Co (at%) | O (at%) | f. u. (±0.01) |
|--------|----------|----------|----------|--------------|
| La:Co  | 21.14(2) | 20.08(5) | 58.77(3) | LaCo$_{0.95}$O$_{2.78}$ |
| Excess Co | 20.36(6) | 20.36(4) | 59.26(6) | LaCo$_{0.92}$O$_{2.92}$ |
| Excess Co + HP | 19.96(6) | 19.99(6) | 60.00(9) | LaCo$_{3.00}$ |

![Fig. 3](https://example.com/fig3.png)

Fig. 3 (a) The magnetic susceptibilities of studied LaCoO$_3$ samples and the corresponding isothermal magnetizations at 2 K are shown in the inset. (b) Resistivities for the as-prepared and annealed LaCoO$_{3.4}$ are compared, and the inset shows analyzed activation gaps from the Arrhenius plot.
together with those reported in the literature showing Curie-tail but of unknown δ, χ′ can be read directly from T → 0 K for an annealed sample of δ → 0 after the χ′corr correction. The inset shows the thermal activation behavior of a spin gap Q ∼ 24.8 meV. (b) The Curie constant is analyzed to show the thermally activated increase of HS population, which reaches ~50% LS/50% HS in a Boltzmann distribution. (c) The threefold increase in the thermal expansion coefficient α across the same temperature range as the spin gap opening is derived directly from the detected capacitance change due to dilation, as illustrated in the inset.

C. Boltzmann distribution of HS

For LaCoO₃ of δ ~ 0, the 1/|χ|² vs. T linear dependence between ~200–500 K implies a typical PM behavior following the Curie-Weiss law. The Curie-Weiss law fitting of χ(T) = C/(T - Θ) for χVV + χ′(T) data between 200 and 500 K [Fig. 4(a)] used a constant fitting parameter of χVV = 4.2 × 10⁻⁴ cm³ mol⁻¹ read directly from T → 0 K, which satisfactorily returns a Curie constant C = 1.476(5) cm³ mol⁻¹ and a Weiss temperature Θ = −131.7(8) K. The magnitude and sign of Θ corresponds to the minimum thermal energy required to break the antiferromagnetically coupled spins. If we follow the conventional data analysis method by assuming there is only one type of spin specie that is responsible for the PM state described by the Curie-Weiss law, the Curie constant C = N₀μeff²/3kB is contributed from the single spin specie of effective moment μeff = 3.436 μB.

Because the LS(t₁g e₁g), IS(t₂g e₂g), and HS(t₃g e₃g) states have S = 0, 1, and 2, respectively, g = 2.42 for IS and g = 1.39 for HS are calculated from μeff = g√(S(S + 1))μB = 3.436μB under the single specie spin-only assumption. The observation of g > 2 for the first row transition metal element of 3d⁶ filling could be attributed to the incomplete orbital quenching, mostly from the unpaired electrons in the triply degenerate t₂g (dₓ²−dᵧ²+dₓz) orbitals which are operable with 90° orbital rotation. However, the degree of orbital contribution to the effective moment remains unknown theoretically, not to mention that the g < 2 value for HS cannot be explained in this context. In addition, the three-fold degeneracy of t₃g for the octahedral CEF splitting of LaCoO₃ must have been lifted by the pseudo-cubic distortion already. In order to extract more meaningful information from the measured Curie constant, we propose to use a multi-component spin-only model analysis that has been proven successful in the interpretation of the magnetic property of Na₄CoO₂ before.

While the Curie constant represents the sum of all nonzero spins in the PM state without distinguishing the actual spin species of corresponding populations, there are multiple possible distributions of spin size/population that are consistent with the experimental value of

$$\sum N_i μ_i / 3kB$$

under the constraint of

$$N₀ = \sum N_i = N_{LS} + N_{IS} + N_{HS}$$

where the effective moments μᵢ² = g²S(S + 1)μB² (i = 1, 2, 3) correspond to LS, IS, and HS states, respectively. To analyze Curie constant with multiple spin species spin-only model, i.e., g = 2 in the high-temperature PM regime (for T > |Θ| ~ 132 K) of the fully thermally activated stage, the theoretical Curie constants of four possible excited spin distributions for Co³⁺ are calculated to be C = 0.999, 1.999, 2.998, and 1.499 cm³ mol⁻¹ for mixtures of 100% IS, 50% IS/50% HS, 100% HS, and 50% LS/50% HS, respectively, as indicated by the dashed lines shown in Fig. 4(b).
It is interesting to note that the high-temperature saturated value of the Curie constant follows the 50% LS/50% HS distribution perfectly. The 50% LS/50% HS statistical distribution at high temperature is illustrated clearly in Fig. 3(b) by the plot of the temperature dependence of the Curie constant \( C(T) = N(T) \mu_a e^2/3k_B = \chi(T)(T - \Theta) \) derived from the Curie–Weiss behavior of \( \chi(T) \), where

\[
\frac{N(T)}{N_0} = \frac{N^{HS}}{N^{HS} + N^{LS}} \frac{1}{2}
\]

above \( \sim 300 \) K; i.e., \( N^{HS} = N^{LS} \approx 1 \) when \( T > 300 \) K with thermal energy of \( k_B T > 0 \sim 25 \) meV. Although any other transient intermediate state like the IS state cannot be ruled out completely without spectroscopic evidence, the nearly perfect demonstration of thermally activated Boltzmann distribution of a two-level spin system strongly suggests that the spin state of \( \text{Co}^{3+} \) fluctuates between the LS-HS states statistically. Above all, even when the supplied thermal energy is larger than the gap size, a thermodynamic system must reach its equilibrium state of Boltzmann statistical distribution, instead of activating all spins to the excited state all at once under the consideration of enthalpy reduction only, because the increased entropy can lower the Gibbs free energy through \(-T\Delta S\) as well.

The single spin state assumption without considering the statistical nature of Boltzmann distribution could lead an error of unphysical spin-only moment and overpopulation for the excited state.

\[ \text{D. Phonon softening and crystalline elasticity} \]

Because of the close relationship between spin state and crystal lattice, there have been a number of detailed investigations reported on the structural properties of nominally stoichiometric \( \text{LaCoO}_3 \) using temperature-dependent X-ray and neutron diffraction techniques with the aim of obtaining a more precise determination of this relationship. Instead of using a diffraction technique for the averaged lattice parameters, we have extracted the linear thermal expansion coefficient \( \alpha = (1/L_0)dL/dT \) of a single-crystal sample directly with a newly designed capacitance-type device incorporated into the PPMS, as illustrated schematically in the inset of Fig. 4(c). \( L(T) \) data were extracted from measurements of the capacitance \( C_0 = \varepsilon_0 A/L \) between 5 and 300 K, which are consistent with the lattice parameters obtained from high-resolution neutron diffraction data with \( L_0 \) calibrated to the room-temperature lattice parameter. The \( \alpha \) values calculated from the slope of \( L(T) \) for \( T \geq 135 \) K (\( \alpha_1 \sim 26.5 \times 10^{-6} \) K\(^{-1}\)) are nearly three times larger than those extracted from \( T \leq 30 \) K (\( \alpha_1 \sim 10.1 \times 10^{-6} \) K\(^{-1}\)). It is also interesting to see that the variation of \( \alpha(T) \) is consistent with the temperature dependence of \( \chi(T) \) [Fig. 4(a) and Table 2] in three ranges: \( \alpha \) is low for \( T \leq 30 \) K, corresponding to the \( T \)-independent \( \chi^\text{VV}; \alpha \) is high between Weiss temperature \( |\Theta| \sim 132 \) K and Debye temperature \( \theta_D \sim 512 \) K, corresponding to the PM behavior of HS spins saturating to the 50% LS/50% HS level; and \( \alpha \) behaves nonlinearly between \( \sim 30 \) and \( |\Theta| \sim 132 \) K, corresponding to the growing thermally activated HS population. A strong spin–orbit–phonon coupling is hinted on crossing the temperature range of thermally activated phonon softening.

The bond length of \( \text{Co}^{3+}(\text{LS}+\text{HS})_6 \) fluctuation had been observed by Zhang et al. with resonance ultrasound spectroscopy. Their discussion about the relaxation processes in terms of thermally activated Debye-like freezing processes is exactly in line with our proposed model for interpretation in this study. Although most of the early works treated the spin states of LS and HS to be a simple thermal activation process for the electrons excited to the unoccupied CEF state, Zhang et al. had examined elastic and inelastic oscillation modes to demonstrate not only spin–spin relaxation but also spin-lattice relaxation dynamically and randomly. The proposed complex local Jahn–Teller deformation with concomitant local FM field is consistent with our proposed symmetry breaking related to the \( \text{Co}^{3+} \) size variation in time and positions. Although the thermal expansion coefficient \( \alpha \) goes down to zero at 0 K theoretically without considering the quantum nature of the solid state, the observation of \( \alpha \) changes three times in the two regimes in our experiments is consistent with the high resolution synchrotron X-ray results reported by Bull and Knight.

\[ \text{E. Magnetic specific heat contribution from HS} \]

The specific heats of \( \text{LaCoO}_3 \) between 2 and 300 K were measured with the PPMS using a pulsed heating method, and those between 300 and 600 K were measured with a differential scanning calorimeter (DSC) (STA 449F3, NETZSCH), as shown in Fig. 5(a). The Dulong–Petit limit \( 3RZ = 3 \times 8.314 \) J (mol K\(^{-1}\)) \times 5 = 124.71 J (mol K\(^{-1}\)) \( Z = 5 \) for \( \text{LaCoO}_3 \) is approached above \( \sim 500 \) K to be near the Debye temperature \( \theta_D \). The total specific heat includes electronic \( (C_p^\text{el}) \), phonon \( (C_p^\text{ph}) \), and magnetic \( (C_p^{\text{mag}}) \) contributions. Although no long-range magnetic ordering has been identified in \( \text{LaCoO}_3 \), except for the spin-gapped ground state below \( \sim 30 \) K, the magnetic contribution \( C_p^{\text{mag}}(T) \) in the PM regime should not be excluded, because each spin in the excited HS state must carry magnetic energy higher than its ground LS state. Since the phonon contribution \( C_p^\text{ph}(T) \) is expected to be reduced after the phonon is softened above \( \sim 30 \) K, the loss of \( C_p^\text{ph}(T)(\Delta C_p^\text{ph}(T)) \) could be transferred to

| Table 2 | Strongly correlated crystalline, magnetic, and thermodynamic behaviors of \( \text{LaCoO}_3 \) |
|---|---|---|---|---|
| \( T \leq 30 \) K | \( 30 \leq T \leq |\Theta| \sim 132 \) K | \(|\Theta| \leq T \leq \theta_D\) | \( T \geq \theta_D \sim 521 \) K |
| \( \alpha(T) \) | \( \alpha_1 \sim 10.1 \times 10^{-6} \) K\(^{-1}\) | Nonlinear | \( \alpha_2 \sim 26.5 \times 10^{-6} \) K\(^{-1}\) |
| \( \chi(T) \) | \( \chi^\text{VV} + \chi^\text{VV} \) | Boltzmann distribution | \( \chi(T) \sim C(T - \Theta) \) |
| HS% | \( \sim 0 \) | \( \sim 50\% \text{LS}/50\% \text{HS} \) |
| \( C_p \) | \( C_p^\text{el} + C_p^\text{ph}(T) \) | Phonon softening | \( C_p^\text{ph}(T) + [C_p^\text{ph} - \Delta C_p^\text{ph}(T)](T) \) |

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be estimated by subtracting the “Debye model constructed \( C_p^{\text{ph}}(T) \) before phonon softening and without magnetic contribution” from the experimental total \( C_p(T) \).

It should be noted that each data point of \( C_p^{\text{mag}}(T) \) described by the Debye model is non-analytical and can only be obtained through numerical evaluation of the Debye integral using \( \theta_D/T \) as the upper bound, and the only parameter \( \theta_D \sim 521 \text{ K} \) is derived from \( \beta \sim 12\pi^4 R Z / 5 \theta_D^3 \) for \( T \leq 10 \text{ K} \) before phonon softening. The phonon contribution of \( C_p^{\text{ph}}(T) \) at each temperature could be described as

\[
C_p^{\text{ph}}(T) = 3 R Z \left[ \frac{3}{x^4} \int_0^x \frac{x^4 e^x dx}{(e^x - 1)^2} \right]
\]

where \( R = N_A k_B \), \( x = h v / k_B T \), and \( x_\theta = h v / \theta_D T = \theta_D / T \). A complete \( C_p^{\text{ph}}(T) \) spectrum without phonon softening, and thus without \( C_p^{\text{mag}}(T) \) contribution, is constructed up to 300 K.

\( C_p^{\text{el}}(T) + C_p^{\text{ph}}(T) \) calculated using the low-temperature estimate of \( \gamma \) and \( \beta(\theta_D) \) is shown together with the experimental total \( C_p(T) \) in the inset of Fig. 4(b). Although LaCoO\(_3\) has been categorized as an insulator according to its large calculated band gap and the value of \( \gamma \) seems negligible, \( C_p^{\text{mag}}(T) \sim \gamma T \) cannot be ignored in the high \( T \) range within the Fermi–Dirac statistics [see inset of Fig. 5(a)], otherwise the magnitude of \( C_p^{\text{mag}}(T) \) would be underestimated. As shown in Fig. 5(b), a significant difference is found between the calculated \( C_p^{\text{mag}}(T) + C_p^{\text{ph}}(T) \) without magnetic contribution and the experimental values of total \( C_p(T) \) for \( T \leq 30 \text{ K} \), which can be attributed to the loss of \( C_p^{\text{mag}}(T) \) after phonon softening, or equivalently, to the gain of \( C_p^{\text{mag}}(T) = \Delta C_p^{\text{mag}}(T) \) as a result of strong spin–orbit–phonon coupling. The magnetic internal energy \( U^{\text{mag}}(T) \) can then be evaluated from \( C_p^{\text{mag}}(T) \) through the integral of

\[
U^{\text{mag}} = \int C_p^{\text{mag}}(T) dT
\]

as shown in Fig. 5(c).

It is gratifying to find that a similar temperature dependence is found for the HS population growth from the \( \chi(T) \) data analysis independently [Fig. 4(b)], as if the total \( U^{\text{mag}} \) were coming from the population growth of spins at the HS state carrying a fixed amount of excited energy, i.e., \( \Delta E \) between \( \epsilon_g - \Delta \).

We may trace the magnitude of \( \Delta E \) using the magnitude of \( U^{\text{mag}}(300 \text{ K}) \) from the population of HS (50% LS/50% HS at 300 K) based on the assumption that each Co\(^{3+}\) at HS state contributes \( \Delta E \) to the magnetic system. \( \Delta E \) is estimated to be \( \sim 0.488 \text{ meV} \) from \( U^{\text{mag}}(300 \text{ K}) = 23.5 \text{ J mol}^{-1} \text{ via}

\[
U^{\text{mag}} = 1/2 N_A (\text{mol}^{-1}) \times \Delta E (\text{eV}) \times 1.6 \times 10^{-19} (\text{J eV}^{-1})
\]

which is in excellent agreement with the value of activation energy \( \Delta E = 0.5 \text{ meV} \) fitted from the Schottky-type \( C_p^{\text{mag}}(T) \) data analysis before,\(^{38}\) although it was viewed “striking” and puzzling by assuming being solely contributed from the \( \sim 0.1\% \) Co at \( T \leq 10 \text{ K} \). As a supporting evidence for the identification of such a narrow gap, we find that a similar value of \( \Delta E \sim 0.6 \text{ meV} \) has also been obtained from the zero-field splitting energy of \( D = 4.9 \text{ cm}^{-1} \) detected using an electron spin resonance (ESR) technique.\(^{39}\)
induced form of Debye model nearly perfectly. The universality of scaled reduced C\textsubscript{p} values collapsed on the theoretical C\textsubscript{p}(T)/R vs. T/\theta\textsubscript{D} model, which covers samples of chemical bonding types ranging from ionic, metallic, to covalent bondings. We have plotted the experimental C\textsubscript{p}(T) data of LaCoO\textsubscript{3} in the scaled form of C\textsubscript{p}(T)/R vs. T/\theta\textsubscript{D} (\theta\textsubscript{D} = 521 K) as shown in Fig. 6. The universality is clearly demonstrated by the collapsed data points of several compounds with different chemical bonding types after scaling, which also overlap with the scaled theoretical Debye model nearly perfectly. The universality of scaled reduced temperature T/\theta\textsubscript{D} also suggests that the phonon softening induced \(\Delta C\textsubscript{ph}^p\) to \(C\textsubscript{ph}^\text{mag}\) transfer is absorbed by the parameter of \(\theta\textsubscript{D}\) completely as \(\theta\textsubscript{D} > 0\) (see Table 2), which is consistent to the definition of \(\theta\textsubscript{D}\) as the upper bound of the normal mode for lattice vibration.\textsuperscript{37} It is also noteworthy to find that the ratio of Debye temperature (\(\theta\textsubscript{D}\)) to melting point (m.p.) for LaCoO\textsubscript{3} (\(\theta\textsubscript{D},\text{m.p.} = 521/1750 \sim 0.3\)) is only half of that for diamond as a typical rigid covalent compound (\(\theta\textsubscript{D},\text{m.p.} = 2230/3700 \sim 0.6\)), which once again reacts the unique elastic nature of crystallinity for LaCoO\textsubscript{3}.

**F. Band gaps and spin gaps**

The resistivity as a function of temperature for two representative nonstoichiometric and stoichiometric samples are compared in Fig. 3(b). The onset of sharp resistivity increase as a signature of metal-to-insulator transition for sample with oxygen vacancies is reduced significantly, presumably as a result of the carrier doping created by the oxygen vacancy. According to the Arrhenius plot shown in the inset of Fig. 3(b), besides the high temperature larger activation gap of \(~400\) meV extracted between 200–400 K for both, the sample with oxygen vacancy shows additional activation gap of \(~150\) meV at lower temperature between 80–200 K, which is consistent to a band picture of larger intrinsic CB-VB band gap \(~400\) meV plus a smaller gap \(~150\) meV across the impurity band above VB (p-type doping) or below CB (n-type doping). Based on the Seebeck measurement results (not shown),\textsuperscript{41} the carrier doping from oxygen vacancy is confirmed to be p-type, and a schematic band picture is proposed as shown in Fig. 7. The band gap near \(~150\) meV is close to that reported by English et al. on a sample having similar Curie-tail behavior.\textsuperscript{42} On the other hand, the larger gap of \(~400\) meV is consistent to the optical gap size extracted from the synchrotron X-ray reflectivity spectrum and that calculated using the density functional theory (DFT) under GGA+U approximation.\textsuperscript{43,44} Mostly importantly, the oxygen vacancy-free sample shows single gap of \(~400\) meV, in contrast to the sample having oxygen vacancy to show two gaps, including the intrinsic CB-VB gap of \(~400\) meV and an additional smaller gap between IB-VB of \(~150\) meV.

Besides the band gaps extracted from the thermal activation behavior of resistivity shown above, two more narrow gaps, including the CEF gap \(\Delta\text{E} \sim 0.5\) meV estimated from the magnetic internal energy \(U\text{mag}\) gain [Fig. 5(c)] due to increasing HS population, and the spin gap \(\sim 25\) meV extracted from the thermal activation behavior of \(x(T)\) [Fig. 4(a)], are closely related.

![Fig. 6](image.png) The reliability of phonon specific heat estimated with the Debye model is confirmed by the scaling of the experimental data of LaCoO\textsubscript{3} in \(C\textsubscript{p}(T)/R\) vs. \(T/\theta\textsubscript{D}\), where several representative compounds of different chemical bonding types (reproduced following ref. 43) collapse onto the same scaling relationship.

![Fig. 7](image.png) A proposed band picture of LaCoO\textsubscript{3} with an oxygen-deficiency induced p-type impurity band (IB) between the valence band (VB) and conduction band (CB).

| Table 3 | Summary of energy gaps identified in LaCoO\textsubscript{3} including the CEF gap \(\Delta\text{E}\) from \(C\textsubscript{p}(T)\), the thermally activated spin gap \(Q\) from \(x(T)\), and the band gaps \(E\textsubscript{p}\) from resistivity data [Fig. 3(b)] of this work. Several other values reported in the literature are compared |
|---|---|---|---|---|
| \(\Delta\text{E}\) | \(Q\) | \(E\textsubscript{p}\) (<80 K) | \(E\textsubscript{p}\) (80–200 K) | \(E\textsubscript{p}\) (200–400 K) |
| LaCoO\textsubscript{3} | 0.5 | 25 | 405 |
| LaCoO\textsubscript{3}\textsubscript{0.92} | | 153 | 400 |
| LaCoO\textsubscript{3–0.5} | 38\textsuperscript{a} | 146\textsuperscript{a} | 340\textsuperscript{b}, 310\textsuperscript{c} |

\textsuperscript{a} Ref. 45 from resistivity measurement. \textsuperscript{b} Ref. 46 from impedance analysis. \textsuperscript{c} Ref. 48 from DFT calculations.
to the quantum nature being reflected on the spin state. The band gaps and spin gaps deducted from this work and those reported in the literature are summarized in Table 3. The difference between $\Delta E$ and $Q$ is not just one order of magnitude on size, the former is so small that allows spontaneous HS excitation of quantum nature even for $T \rightarrow 0$ K. The narrow $\Delta E$ is also responsible for the temperature-independent paramagnetic contribution of $\chi^{VV}$ as an orbital Zeeman effect for the whole temperature range, mostly because the dominant contribution of $\chi^{VV}$ must be coming from the narrow CEF gap $\Delta E$ of the energy spectrum in the denominator of the second order perturbation for $\chi = \partial^2 E / \partial H^2$ as

$$\chi^{VV} \propto \sum_{n \neq 0} |\langle n | \mathcal{H} | 0 \rangle|^2 / (E_n - E_0)$$

where $\mathcal{H}$ is the Zeeman operator. In particular, the thermally activated gap $Q$ must have little contribution to $\chi^{VV}$ for its much larger gap size and thus little contribution to the perturbation.

It must be stressed that the size of actual CEF-induced energy splitting is often unknown theoretically, let alone its additional fine splitting due to the octahedral CEF distortion, e.g., the Jahn–Teller effect. Current work using LaCoO$_3$ sample without low temperature Curie-tail behavior provided a unique opportunity to obtain a reliable $\chi^{VV}$ value which is contributed from the quantum fluctuation across the extremely narrow CEF gap $\Delta E$. On the other hand, the thermally activated gap $Q$ is reflected on the HS population growth with the assistance of phonon softening mechanism, which indicates energy transfer from the external thermal energy to the system magnetic energy by way of spin–orbit–phonon coupling, as also reflected on the tripled thermal expansion coefficient $a(T)$ across the same temperature range [see Fig. 4(c)]. We might view the phenomenon of phonon softening as a prerequisite for the HS population growth that requires longer (Co–O)$_{HS}$ bond length.

G. EELS study: why is the CEF gap so small?

Because the CEF splitting for Co$^{3+}$ ion sitting in the center of a slightly distorted CoO$_6$ octahedral environment remains unknown and has only been deduced to be near 1 eV, the actual CEF-induced $e_g$–$t_{2g}$ splitting is not likely to be any one of the gaps summarized in Table 3. However, we argue that the actual $e_g$–$t_{2g}$ splitting corresponds to the narrow gap of $\Delta E \sim 0.5$ meV for LaCoO$_3$, and the thermal activation gap $Q$ for HS excitation is achieved through thermal-strain-magnetic energy exchange with the help of a strong spin–orbit–phonon coupling after the phonon softening. It is reasonable to interpret the Hund’s rule for the HS configuration that two electrons are excited to the $e_g$ level so that the Coulomb repulsion among electrons in 3d-orbitals is minimized (see Fig. 8). The softened phonon of expanded average interatomic distance is preferable for the HS state of larger ionic size, which has been confirmed by the study thermal expansion as shown in Fig. 4(c). This direction of interpretation basically follows the developmental concepts of small polaron and the breathing type distortion of

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**Fig. 8** The schematic plot of 3d orbitals split by the octahedral crystal electric field, (a) without La–O covalency, and (b) with La–O covalency that leads to a more isotropic charge distribution and thus a smaller CEF gap.
ligands consistently. The immediate question following would be: why is the CEF splitting (ΔE) so small for LaCoO₃?

Since the origin of an octahedral CEF splitting is coming from the static electric field created by the uneven electron distribution of six ligands surrounding the Co center, a more isotropic environment of negative effective charge of ligands is expected to reduce the CEF size significantly. On examining the ligands surround the Co center of LaCoO₃, although Krapek et al. have proposed that Co–O covalency is responsible for the size reduction of the effective moment for the HS state, the covalency of Co–O is not sufficient to assist a more isotropic negative charge distribution, as illustrated by the six dp hybridized orbitals still pointing toward the six neighboring oxygens (see Fig. 8). The most likely source to create less effective negative charge of oxygen ligands is that partial valence electrons of O²⁻ is pulled away from the region along the six Co–O bonding directions and re-distribute more evenly toward the eight La–O bonding directions, as illustrated in Fig. 8(b). It is likely that the La–O bonding is different to the more ionic nature of Na–O and Sr–O in NaCoO₂ and SrCoO₃. Therefore, the La–O covalency may not be ignored anymore in LaCoO₃. In the study of LaCoO₃, the impact of Sr substitution to the La site has been examined repeatedly in the literature. However, the possibility of Sr substitution induced oxygen vacancy has often been ignored, which complicates the Co spin state interpretation due to the intertwined factors among local strain, electron count, and vacancy ordering. In order to examine the possibility of La–O covalency, the EELS spectrum of LaCoO₃ has been explored for the evidence of plasmonic resonance coming from the paired covalent electrons among La–O.

The STEM-EELS study results of LaCoO₃ are shown in Fig. 9. In contrast to the X-ray photoemission spectroscopy (XPS) which reveals the information of the excitation from the occupied states of valence band, the electron energy-loss spectroscopy (EELS) can provide the information of excitation from both the valence and conduction bands. The EELS spectrum contains the excitations of valence electrons in collective modes, including surface and volume plasmons, single-particle excitations, inter-band transitions, and low-lying core-level ionizations. Fig. 9(a) and (c) shows the low magnification and atomic resolved high-angle annular dark field (HAADF) images in scanning TEM (STEM), respectively. Fig. 9(b) shows the electron diffraction pattern correlated to Fig. 9(a). Indeed, the atomic resolved HAADF image recorded along [001] direction revealed a pseudo-cubic symmetry which is consistent with literature report. Although the ferroelastic domain walls in LaCoO₃ have widely observed in literature, however, our HP-LaCoO₃ single crystal exhibits high crystal quality including the crystallinity and domain-free in this study. The EELS spectrum for LaCoO₃ [Fig. 9(f)] reveals several spectral features of incident beam energy loss at ~13.9, ~30.4 eV, and the ~63.3 eV (not shown) from Co M₂,₃ edge excitation. Within the framework of macroscopic dielectric-response theory, we derived the frequency-dependent dielectric function (ε = ε₁ + iε₂) of LaCoO₃ by performing the Kramers–Kronig Analysis (KKA) as shown in Fig. 9(g). Since ε₁ passes through zero at ~13.9 and ~30.4 eV with accompanied decrease of ε₂, which gives rise to the maxima of the volume loss function Im(1/ε(ω)) and the absorptions near ~13.9 and ~30.4 eV show characteristics of volume-plasmon excitations and also confirm with highest intensity localized in bulk interior in Fig. 9(d) and (e), respectively, performed by spectrum imaging. The physical origin of the volume plasmons near ~13.9 and ~30.4 eV are proposed coming from the collective plasmonic resonance of valence electrons among La–O and Co–O, respectively. The EELS evidence of nontrivial La–O covalency supports the scenario that the reduced and isotropic effective negative charge of oxygen ligands is responsible for the significantly reduced octahedral CEF gap size of ΔE ~ 0.5 meV for LaCoO₃.

Fig. 9 The STEM-EELS study results of LaCoO₃. (a)–(c) The scanned area, diffraction pattern, and atomic scale real lattice image of LaCoO₃ along one of the axes of pseudo-cubic symmetry. (d)–(e) The EELS map of two specific energy absorption windows near ~13–15 and 30–32 eV. (f) The EELS spectrum of LaCoO₃ shows energy absorption of two volume plasmons corresponding to the collective resonance of σ-bond electrons of La–O and Co–O, respectively. (g) The real and imaginary part of the dielectric constant (ε = ε₁ + iε₂) with a blow-up view to show that ε₁ passes zero.

H. The implication of crystalline elasticity

It is reasonable to assume that the bond length of (Co–O)HS is slightly longer than that of (Co–O)HS for electron filling in the higher ε₈ energy level of HS (t₂g²e₈¹) state, which could lead to direct correlation between the magnetic and phonon energies via spin–orbit–phonon coupling, as shown in Fig. 4 and the published high resolution lattice parameters. In particular, when a 3d-electron is excited from t₂g₂ to the higher ε₈ level, concomitant local distortion must occur (see Fig. 10), and
a local breathing mode is established between the magnetic Zeeman energy and the phonon energy of ferroelectric origin. The dilute distribution of HS spins below ~30 K exhibits a low α with a rigid and homogeneous distribution of shorter (Co–O)\textsuperscript{LS} bonds. As the HS population grows with increasing temperature, the initial randomly distributed low population of HS would create high strain to the lattice, until nearly half of the Co ions are excited to the HS state at high temperature.

A more homogeneous magnetic-phonon energy exchange is achieved for the 50% HS/50% LS distribution of minimum lattice strain with well-defined threefold increase of α (see Fig. 4). The reduction of rigidity is prerequisite for the 50%(Co–O)\textsuperscript{HS}/50%(Co–O)\textsuperscript{LS} statistical distribution under thermal fluctuations, which is also reflected on the increase of electron entropy.

Yamaguchi \textit{et al.} have proposed that e\textsubscript{g} orbital ordering (OO) may exist in the Co\textsuperscript{3+} IS state (t\textsubscript{2g}e\textsubscript{g}) of LaCoO\textsubscript{3} having lifted orbital degeneracy in e\textsubscript{g}; i.e., the electron in e\textsubscript{g} prefers to occupy either d\textsubscript{x\textsuperscript{2}}–y\textsuperscript{2} or d\textsubscript{z\textsuperscript{2}} orbital in an ordered pattern (see Fig. 8), which has also been applied to explain the observed split Co–O stretching mode as local distortion arising from the Jahn–Teller effect. Comparing these two proposed model interpretations for the local distortion due to Co–O bond length fluctuation, the controversial IS state could be explained equally well by either the OO of an IS state, or the 50% LS/50% HS statistical distribution in time average. Above all, the distinction between HS and IS is not possible in the high temperature metallic phase anyway.

IV. Conclusion

In conclusion, based on an integrated study of the magnetic susceptibility, specific heat, and thermal expansion of a single crystal LaCoO\textsubscript{3} nearly free of cobalt and oxygen vacancies, we propose a consistent interpretation of strongly correlated spin state and crystal lattice for the pseudo-cubic LaCoO\textsubscript{3} perovskite. The puzzling deviation of the crystal symmetry from cubic to rhombohedral symmetry as a relaxed pseudo-cubic structure is related to the LS/HS spin state transition with corresponding statistically distributed (Co–O)\textsuperscript{LS}–(Co–O)\textsuperscript{HS} bond length fluctuation.

Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

Conflicts of interest

There are no conflicts to declare.

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References

1 H. L. Yakel, Acta Crystallogr., 1955, 8, 394.
2 W. C. Koehler and E. O. Wollan, J. Phys. Chem. Solids, 1957, 2, 100.
3 M. A. Senaris-Rodriguez and J. B. Goodenough, J. Solid State Chem., 1995, 116, 224.
4 M. W. Haverkort, et al., Phys. Rev. Lett., 2006, 97, 176405.
5 S. W. Biernacki, Phys. Rev. B: Condens. Matter Mater. Phys., 2006, 74, 184420.
6 M. A. Korotin, S. Y. Ezhov, I. V. Solovyev, V. I. Anisimov, D. I. Khomskii and G. A. Sawatzky, Phys. Rev. B: Condens. Matter Mater. Phys., 1996, 54, 5309.
7 S. Yamaguchi, Y. Okimoto and Y. Tokura, Phys. Rev. B: Condens. Matter Mater. Phys., 1997, 55, R8666.
8 P. G. Radaelli and S.-W. Cheong, Phys. Rev. B: Condens. Matter Mater. Phys., 2002, 66, 094408.
9 C. Zobel, M. Kiener, D. Bruns, J. Baier, M. Gruninger, T. Lorenz, P. Reuter and A. Revcolevschi, Phys. Rev. B: Condens. Matter Mater. Phys., 2002, 66, 020402.
10 G. Vanko, J. P. Rue, A. Mattila, Z. Nemeth and A. Shukla, Phys. Rev. B: Condens. Matter Mater. Phys., 2006, 73, 024424.
11 S. N. Thant, K. Toshiaki, K. Yoshihiko, S. Masaru and A. Kichizo, J. Phys. Soc. Jpn., 2006, 75, 084601.
12 R. Schmidt, J. Wu, C. Leighton and I. Terry, Phys. Rev. B: Condens. Matter Mater. Phys., 2009, 79, 125105.
13 G. Maris, Y. Ren, V. Volotchaev, C. Zobel, T. Lorenz and T. T. M. Palstra, Phys. Rev. B: Condens. Matter Mater. Phys., 2003, 67, 224423.
14 Y. Kobayashi, T. S. Naing, M. Suzuki, M. Akimitsu, K. Asai, K. Yamada, J. Akimitsu, P. Manuel, J. M. Tranquada and G. Shirane, Phys. Rev. B: Condens. Matter Mater. Phys., 2005, 72, 174405.
15 R.-P. Wang, A. Hariki, A. Sotnikov, F. Frati, J. Okamoto, H.-Y. Huang, A. Singh, D.-J. Huang, K. Tomiyasu, C.-H. Du, J. Kunes and F. M. F. de Groot, Phys. Rev. B: Condens. Matter Mater. Phys., 2018, 98, 035149.
16 J. B. Goodenough, J. Phys. Chem. Solids, 1958, 6, 287.
17 J. B. Goodenough, Czech J. Phys., 1967, 17, 304.
18 J. B. Goodenough, Phys. Rev., 1967, 164, 785.
19 J. B. Goodenough, J. Appl. Phys., 1966, 37, 1415.
20 N. Biskup, J. Salafranca, V. Mehta, M. P. Oxley, Y. Suzuki, S. J. Pennycook, S. T. Pantelides and M. Varela, Phys. Rev. Lett., 2014, 112, 087202.
21 K. Asai, A. Yonedo, O. Yokokura, J. M. Tranquada, G. Shirane and K. Kohn, J. Phys. Soc. Jpn., 1998, 67, 290.
22 R. F. Egerton, Electron Energy-Loss Spectroscopy in the Electron Microscope, Springer Science, 2011.
23 J. –Q. Yan, J. –S. Zhou and J. B. Goodenough, Phys. Rev. B: Condens. Matter Mater. Phys., 2004, 70, 014402.
24 B. Pimentel, R. J. Carballo-Vivas, N. R. Checa, D. L. Rocco and M. S. Reis, J. Alloys Compd., 2019, 796, 331–337.
25 E. du Tremolet de Lacheisserie, D. Gignoux, and M. Schlenker, Magnetism: II-Materials and Applications, Springer US, 2012.
26 T. Kyomen, Y. Asaka and M. Itoh, Phys. Rev. B: Condens. Matter Mater. Phys., 2005, 71, 024418.
27 M. Itoh, M. Sugahara, I. Natori and K. Motoya, J. Phys. Soc. Jpn., 1995, 64, 3967.
28 K. Asai, P. Gehring, H. Chou and G. Shirane, Phys. Rev. B: Condens. Matter Mater. Phys., 1989, 40, 10982.
29 K. Knizek, Z. Jirak, J. Hejmanek and P. Novak, J. Phys.: Condens. Matter, 2006, 18, 3285.
30 R. Marx and H. Happ, Phys. Status Solidi B, 1975, 67, 181.
31 G. J. Shu, J. C. Tian, C. K. Lin, M. Hayashi, S. C. Liou, W. T. Chen, D. P. Wong, H. L. Liou and F. C. Chou, New J. Phys., 2018, 20, 058002.
32 N. W. Ashcroft and N. D. Mermin, Solid State Phys., 2011, 643–725.
33 G. J. Shu and F. C. Chou, Phys. Rev. B: Condens. Matter Mater. Phys., 2016, 93, 140402.
34 G. J. Shu, S. C. Liou, S. K. Karna, R. Sankar, M. Hayashi and F. C. Chou, Phys. Rev. Mater., 2018, 2, 044201.
35 Z. Zhang, J. Koppensteiner, W. Schranz, D. Prabhakaran and M. A. Carpenter, J. Phys. Condens. Matter, 2011, 23, 14501.
36 C. L. Bull and K. S. Knight, Solid State Sci., 2016, 57, 38.
37 C. Kittel, Introduction to Solid State Physics, Wiley, 1996.
38 C. He, H. Zheng, J. F. Mitchell, M. L. Foo, R. J. Cava and C. Leighton, Appl. Phys. Lett., 2009, 94, 102514.
39 S. Noguchi, S. Kawamata, K. Okuda, H. Nojiri and M. Motokawa, Phys. Rev. B: Condens. Matter Mater. Phys., 2002, 66, 094404.
40 N. Manini, Introduction to the Physics of Matter, Springer, 2015.
41 M. Dragan, S. Enache, M. Varlam and K. Petrov, in Cobalt Compounds and Applications, IntechOpen, 2019.
42 S. R. English, J. Wu and C. Leighton, Phys. Rev. B: Condens. Matter Mater. Phys., 2002, 65, 220407.
43 E. Iguchi, K. Ueda and W. H. Jung, Phys. Rev. B: Condens. Matter Mater. Phys., 1996, 54, 17431.
44 T. Arima, Y. Tokura and J. B. Torrance, Phys. Rev. B: Condens. Matter Mater. Phys., 1993, 48, 17006.
45 S. Zhang, N. Han and X. Tan, RSC Adv., 2015, 5, 760.
46 R. M. White, Quantum theory of magnetism, Magnetic Properties of Materials, Springer Berlin Heidelberg, Berlin, Heidelberg, vol. 32, 2007.
47 D. C. Johnston, Phys. Rev. Lett., 1989, 62, 957.
48 P. M. Raccah and J. B. Goodenough, Phys. Rev., 1967, 155, 932.
49 V. Krapek, P. Novak, J. Kunes, D. Novoselov, D. M. Korotin and V. I. Anisimov, Phys. Rev. B: Condens. Matter Mater. Phys., 2012, 86, 195104.
50 G. J. Shu and F. C. Chou, Phys. Rev. B: Condens. Matter Mater. Phys., 2013, 88, 155130.
51 A. Munoz, C. de la Calle, J. A. Alonso, P. M. Botta, V. Pardo, D. Baldomir and J. Rivas, Phys. Rev. B: Condens. Matter Mater. Phys., 2008, 78, 054404.
52. C. N. R. Rao, M. M. Seikh, and C. Narayana, in *Spin Crossover in Transition Metal Compounds II*, Springer Berlin Heidelberg, Berlin, Heidelberg, 2004, pp. 1–21.
53. H. Raether, *Excitation of Plasmons and Interband Transitions by Electrons*, Springer, 2006.
54. R. F. Klie, J. C. Zheng, Y. Zhu, M. Varela, J. Wu and C. Leighton, *Phys. Rev. Lett.*, 2007, **99**, 047203.
55. J. C. Walmsley, A. Bardal, K. Kleveland, M. –A. Einarsrud and T. Grande, *J. Mater. Sci.*, 2000, **35**, 4251–4260.
56. N. Orlovskaya, N. Browning and A. Nicholls, *Acta Mater.*, 2003, **51**, 5063–5071.
57. S. C. Liou, M.-W. Chu, R. Sankar, F.-T. Huang, G. J. Shu, F. C. Chou and C. H. Chen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 085126.
58. B. Chakrabarti, T. Birol and K. Haule, *Phys. Rev. Mater.*, 2017, **1**, 064403.