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Carrier-specific dynamics in 2H-MoTe₂ observed by femtosecond soft x-ray absorption spectroscopy using an x-ray free-electron laser

ABSTRACT

Femtosecond carrier dynamics in layered 2H-MoTe₂ semiconductor crystals have been investigated using soft x-ray transient absorption spectroscopy at the x-ray free-electron laser (XFEL) of the Pohang Accelerator Laboratory. Following above-bandgap optical excitation of 2H-MoTe₂, the photoexcited hole distribution is directly probed via short-lived transitions from the Te 3d⁵/₂ core level (M₅-edge, 572–577 eV) to transiently unoccupied states in the valence band. The optically excited electrons are separately probed via the reduced absorption probability at the Te M₅-edge involving partially occupied states of the conduction band. A 400 ± 110 fs delay is observed between this transient electron signal near the conduction band minimum compared to higher-lying states within the conduction band, which we assign to hot electron relaxation. Additionally, the transient absorption signals below and above the Te M₅ edge, assigned to photoexcited holes and electrons, respectively, are observed to decay concomitantly on a 1–2 ps timescale, which is interpreted as electron–hole recombination. The present work provides a benchmark for applications of XFELs for soft x-ray absorption studies of carrier-specific dynamics in semiconductors, and future opportunities enabled by this method are discussed.

INTRODUCTION

The optically excited carrier relaxation and structural dynamics of semiconductors govern their optoelectronic properties and functionality in emerging device applications. Femtosecond time-resolved x-ray spectroscopy and scattering are powerful techniques to track the electronic and structural dynamics of such materials in real time. With the recent development of ultrafast x-ray and extreme
ultraviolet (XUV) sources using x-ray free electron lasers (XFELs) and high-harmonic generation (HHG), new opportunities are being explored for investigating charge-carrier dynamics in the condensed phase with carrier-, element-, and oxidation-state specificity. The ability to probe the valence electronic structure via localized core levels using x-ray/XUV absorption spectroscopy (XAS) has been exploited to capture electron and hole carrier relaxation dynamics separately in bulk semiconductors and to measure layer-specific dynamics within multi-component heterojunctions. These early successes have been led primarily by applications of table-top HHG sources for XUV transient absorption and reflection spectroscopy.

With the recent emergence of XFELs, which produce femtosecond x-ray pulses with ~6–8 orders of magnitude greater spectral brightness per pulse compared to HHG, a new horizon is coming into view for ultrafast core-level spectroscopy of semiconductors. In addition to the potential for detecting weak signals using a high-XFEL photon flux, the small spatial scale (~1–10s μm) of the focused x-ray pulses, and the much longer penetration depths of higher-energy x-rays allow for smaller and thicker semiconductor samples, respectively, to be investigated. Furthermore, the small spin–orbit energy splittings of elemental core levels within the XUV range, which are often comparable to the bandgap of relevant semiconductors (1–3 eV), lead to overlapping spectral features that frequently complicate the interpretation of hole and electron dynamics in XUV spectroscopy. Although deconvolution of the spin–orbit split transitions has been possible in some examples, the overlapping spin–orbit features can be completely eliminated by using the higher-energy x-rays from XFELs at deeper core-level edges that are well-separated compared to the bandgap energies. In the present work, we apply femtosecond optical-pump, x-ray-probe spectroscopy at the Te M5 edge (572–577 eV) using an XFEL to capture the carrier-specific dynamics of a prototypical layered semiconductor, 2H-MoTe2. Our study provides a benchmark for XFEL-based soft x-ray femtosecond transient absorption spectroscopy of semiconductor materials.

MoTe2 is a transition metal dichalcogenide (TMDC) within the class of layered materials, like graphite, that are composed of two-dimensional sheets (monolayers) bound by weak van der Waals forces. In contrast to graphite and its corresponding monolayer form, MoTe2 is stable in the bulk and as a monolayer in both a semimetallic phase (1T) and semiconductor phase (2H). The 2H-phase MoTe2 has a bandgap of ~0.9 eV in the bulk and 1.1 eV in the monolayer form, which are similar to that of silicon (1.1 eV). Various applications based on both monolayer and multilayer MoTe2 have been investigated, including field-effect-transistors, photonic logic gates, and phase-change devices. The advancement of next-generation optoelectronics based on layered TMDC materials relies on a detailed understanding of the relaxation and transport of the carriers. Therefore, recent studies using ultrafast optical and THz absorption spectroscopies have examined carrier lifetimes in 2H-MoTe2. However, these measurements lack both carrier-specificity and sensitivity to hot electron and hole dynamics prior to electron–hole recombination or trapping.

Time-resolved XAS can distinguish electron and hole dynamics in semiconductors, including intraband carrier thermalization and cooling. To achieve this, an optical pump pulse first excites electrons into the conduction band (CB) and creates holes in the valence band (VB); a temporally delayed x-ray pulse then probes the changes in transitions of core-level electrons into the unoccupied states of the VB and CB. The effects of increased core → VB transitions due to the presence of photoexcited holes and decreased core → CB transitions due to excited electrons are collectively referred to as state-filling effects. In a recent work, using XUV transient absorption spectroscopy on 2H-MoTe2, we captured the hot hole scattering dynamics within the VB, the subsequent electron–hole recombination, and coherent lattice dynamics in 2H-MoTe2 induced by strong electron–phonon coupling. The 1.5 eV spin–orbit splitting of the Te M5 and N4 edges, which is comparable to the 2H-MoTe2 bandgap energy (0.9 eV), led to overlapping spectral features, particularly between the hole signal at the N4 edge and the electron state-filling signal at the N4 edge. Unfortunately, this overlap precluded a detailed analysis of the electron thermalization and cooling dynamics.

In the present work, we extend the method established by XUV transient absorption to the x-ray regime at the Te M5 edge of 2H-MoTe2 using an XFEL. We achieve a noise level of ~0.05% in measurements of the x-ray transmission, which surpasses the performance of XUV measurements on the same material and is crucial due to an order-of-magnitude smaller edge jump in absorption at the M5 edge compared to the N4 edge in the XUV. At the Te M5 edge, which is spectrally isolated by >10 eV from neighboring core-level edges, we show that the XAS closely maps the unoccupied density of states (DOS) of the CB in 2H-MoTe2. In this time-resolved optical-pump, XAS-probe experiment, we identify a signature of hot electron cooling within the CB, which was not observed in the recent XUV study due to overlapping state-filling signals from the N4 and N4 edges. In addition, we successfully capture the separate photoexcited hole signal and trace the electron–hole recombination dynamics of 2H-MoTe2.

EXPERIMENTAL METHODS AND CALCULATIONS

Sample preparation and static XAS measurements using synchrotron radiation

The static XAS of both the 1T and 2H phases of MoTe2 are measured in the present work, while pump–probe measurements are carried out only on 2H-phase MoTe2. Samples of 2H- and 1T’-MoTe2 are synthesized via chemical vapor deposition (CVD) and directly deposited on 100 nm thick Si3N4 substrates of 2 × 2 mm2 lateral size. This results in a homogenous thin-film polycrystalline sample for both phases. The structural phase of the samples is confirmed by their Raman spectra (see the supplementary material for more details). The static XAS measurements of the 2H and 1T’ phases of MoTe2 were initially carried out at Beamline 8–2 of the Stanford Synchrotron Radiation Lightsource (SSRL) synchrotron tuned to the Mo M3 and Te M5,4 edges. Further information about soft x-ray absorption spectroscopy at the SSRL can be found in the literature. Several experimental approaches were considered for soft x-ray absorption measurements. In contrast to 3d transition metal L edges, a total or partial fluorescence yield measurement of the Mo and Te M edges is challenging because of their extremely low fluorescence yields of 0.3% and 0.1%, respectively. Thus, either a total electron yield (TEY) measurement or a transmission measurement would be more viable. Since the final goal is to incorporate these XAS measurements into pump–probe studies and the TEY measurement is more prone to artifacts induced by the optical pump laser, we focus in this work on the transmission measurement. To optimize the signal-to-noise ratio, the
MoTe₂ layer thickness during sample synthesis is tailored to obtain approximately one absorption length of the x-rays at the M₅ edge. The resulting 100 nm MoTe₂ layer absorbs ~20% of incident x-rays at energies just below the Te M₅,₄ absorption edges and ~54% above, which results in a change in the transmission by the absorption edge jump of ~34%. However, we note that the change in transmission at the resonant M₅ pre-edge—where the carrier dynamics are probed in the time-resolved measurements—is only 2%.

**Femtosecond time-resolved XAS using XFEL pulses**

The femtosecond time-resolved XAS measurements on 2H-phase MoTe₂ were performed at the Soft X-ray Scattering and Spectroscopy (SSS) beamline of the Pohang Accelerator Laboratory (PAL)-XFEL. A simplified layout of the experiment is shown in Fig. 1; a detailed description of the soft x-ray beamline at PAL-XFEL can be found in the literature. In brief, the x-rays are monochromatized (to a bandwidth of 0.1 eV) and focused onto the sample with a Kirkpatrick–Baez (KB) mirror system, resulting in an x-ray beam diameter of 110 μm × 80 μm at the sample position, thus much smaller than the lateral extent of the thin-film MoTe₂ sample. The monochromatized XFEL energy per pulse without further attenuation is estimated to be 1.6 × 10¹³ carriers/cm² (see the supplementary material for details on the optical absorption at 400 nm). The x-ray transmission photodiode (I₀) is shielded from the transmitted and scattered 400 nm pump light with a free-standing 200 nm thick Al filter. The temporal pump–probe overlap is measured via the x-ray-induced change of 400 nm reflectivity with a single-crystal YAG sample. This measurement was carried out at least twice per 12-h shift and the drift in the centroid of the optical-x-ray temporal overlap was confirmed to be <100 fs on the timescale of a 6- to 12-h measurement. The pump–probe instrument response time is determined to be ~200 fs, as described below.

**ELECTRONIC STRUCTURE CALCULATIONS**

Density functional theory (DFT) with the projector augmented wave (PAW) method³⁵ implemented in the Vienna Ab initio Simulation Package (VASP)³⁶,³⁷ was used to compute the density of states (DOS) for bulk 2H- and 1T'-phases of MoTe₂ crystals without photoexcitation. Exchange and correlation effects were calculated using the Perdew–Burke–Ernzerhof (PBE) form of the Generalized Gradient Approximation (GGA). Wave functions were constructed using a plane wave basis set with components up to kinetic energy of 400 eV and the reciprocal space was sampled using a 3 × 3 × 3 Γ-centered mesh with a 0.05 eV Gaussian smearing of orbital occupancies. DFT simulations of bulk 2H (1T') MoTe₂ crystals were performed on 120-atom supercells measuring 17.58 Å × 12.19 Å × 13.97 Å (17.48 Å × 12.67 Å × 15.43 Å) along the a-, b-, and
c-directions, respectively. Calculations were performed until each self-consistency cycle converged in energy to within $10^{-7}$ eV/atom and forces on ions are under $10^{-2}$ eV/Å.

The calculation of core-level absorption spectra at the Te $M_5$ edge was accomplished with DFT and Bethe–Salpeter equation (BSE) calculations using Quantum ESPRESSO and the Obtaining Core-level Excitations using Ab initio methods and the NIST BSE solver (OCEAN) software package. The DFT-BSE calculation was conducted using norm-conserving scalar-relativistic PBE pseudopotentials with nonlinear core correction under GGA and a $6 \times 6 \times 1$ k-point meshgrid. In the calculation, the number of bands was set to 40 and the dielectric constant was set to 12.9. Convergence was achieved with an energy cutoff of 80 Ryd and a cutoff radius of 4 Bohr. A lifetime broadening of 0.1 eV was assumed in the BSE calculation.

RESULTS AND DISCUSSION

X-ray absorption spectroscopy of 2H- and 1T'-phases of MoTe$_2$

The static XAS of the 2H- and 1T'-phases of MoTe$_2$ thin-film samples at the Te $M_5$-edge acquired at SSRL, are shown in Fig. 2. The spectra cover the distinct $M_3$ and $M_4$ pre-edge features at 573 eV and 584 eV, respectively. The $M_3$ and $M_4$ pre-edge transitions correspond to the promotion of Te 3$d_{5/2}$ and Te 3$d_{3/2}$ core electrons, respectively, to the CB of semiconducting 2H-MoTe$_2$ and to unoccupied states above the Fermi-level in 1T'-MoTe$_2$, as depicted schematically in Fig. 2(b). The broad, atomic absorption to the continuum has its onset at higher energies, starting around the same energy as the $M_4$-edge feature. This delayed onset is due to the centrifugal barrier.

Note the dip in absorption at ~568 eV in the 1T'-MoTe$_2$ spectrum is caused by an imperfect I$_0$ normalization during refill of the synchrotron ring during that measurement.

The inset of Fig. 2(a) shows in more detail the Te $M_5$ pre-edge features of both structural phases of MoTe$_2$. In the semimetallic 1T'-MoTe$_2$, the Te $M_5$ pre-edge onset is red-shifted compared to that of the semiconducting 2H-MoTe$_2$. This is consistent with the bandgap collapse in the 1T' phase compared to the 2H phase. To quantify this effect, we perform DFT calculations of the bulk DOS in both phases and extract the energy of the Fermi level, $E_F$, in the case of bulk 1T'-MoTe$_2$ and of the CB minimum, $E_{CBM}$, in the case of bulk 2H-MoTe$_2$, relative to the vacuum. The corresponding DOS distributions are plotted in Fig. 2(b) and the energy difference between $E_{CBM}(2H)$ and $E_F(1T')$ is 0.36 eV. In the inset of Fig. 2(a), the vertical green dashed line denotes the energy difference between the calculated $E_F(1T')$ and the measured core-level ionization potential of Te 3$d_{5/2}$ (1T'), i.e., the energy of the transition represented by the purple vertical arrow on the left side of Fig. 2(b). The black vertical dashed line marks the energy difference between the calculated $E_{CBM}(2H)$ and the measured core-level ionization potential of Te 3$d_{5/2}$ (2H), i.e., the energy of the transition represented by the purple vertical arrow on the right side of Fig. 2(b). Note that the Te 3$d_{5/2}$ ionization potentials of the two phases are extracted directly from XPS measurements (see the supplementary material, Fig. S2), giving the 0.15 eV difference noted in Fig. 2(b).

\begin{figure}[h]
\centering
\includegraphics[width=\textwidth]{fig2.png}
\caption{(a) Normalized Te $M_{5\alpha}$-edge x-ray absorption spectrum of 2H- and 1T'-phases of MoTe$_2$. Note the dip in absorption at ~568 eV in the 1T'-MoTe$_2$ spectrum is caused by an imperfect I$_0$ normalization during refill of the synchrotron ring during that measurement. The inset shows an expanded plot of the Te $M_5$ pre-edge feature. The green vertical dashed line denotes the energy of the Fermi level, $E_F$, in 1T'-MoTe$_2$ relative to the Te 3$d_{5/2}$ core level of the 1T' phase. The black vertical dashed line denotes the energy of the CB minimum, $E_{CBM}$, in 2H-MoTe$_2$ relative to the corresponding Te 3$d_{5/2}$ core level of the 2H phase. (b) Calculated total DOS of bulk 2H- and 1T'-MoTe$_2$. The dashed horizontal lines mark the calculated energies of $E_F(1T')$ and $E_{CBM}(2H)$ with an energy scale set relative to the Fermi level of 1T'-MoTe$_2$. The vertical purple arrows represent the onset of core-level transitions at the Te $M_5$ edge of both phases, corresponding to the vertical dashed lines in (a). As the x-ray energy is increased above each onset, carriers are promoted to higher-energy states above $E_F(1T')$ and $E_{CBM}(2H)$. Note the slight difference (0.15 eV) in the Te 3$d_{5/2}$ core-level ionization potential in the 1T'-phase compared to the 2H-phase MoTe$_2$ (see the supplementary material for more details), which gives a total difference of 0.51 eV in the expected XAS onsets of the two phases.}
\end{figure}
whereas the energy difference between $E_F(1T^0)$ and $E_{CBM}(2H)$ of 0.36 eV comes from the DFT calculations. The overall expected red shift of the 1T$^0$-MoTe$_2$ $M_5$ pre-edge onset relative to that of 2H-MoTe$_2$ is therefore $0.15 \text{ eV} + 0.36 \text{ eV} = 0.51 \text{ eV}$, which is in good agreement with the experimental XAS. In both phases, although the $M_5$ pre-edge transitions of Te(3d$5/2$) $\rightarrow$ Te(5p) character are dipole allowed, the absorption change (i.e., the edge jump) is weak at these resonances. As seen in Fig. 2(a), the pre-edge features are $\sim$20 times weaker than the continuum resonance of the Te $M_{3,4}$ edge. For a 100 nm thick sample, the x-ray transmission only changes by $\sim$2% at the $M_5$ pre-edge. The clear distinction between the 2H- and 1T$^0$-phase $Te M_{5}$-edge absorption illustrates the sensitivity of XAS to modifications in the valence electronic structure.

In Fig. 3(a), the DFT-calculated band structure of 2H-MoTe$_2$ along the $\Gamma$–$K$–$M$–$\Gamma$ path is plotted. In Fig. 3(b), the corresponding total CB DOS in 2H-MoTe$_2$ is compared to the experimental absorption spectrum of the Te $M_{5}$ pre-edge, after applying a global energy shift to the CB DOS to match the onset of the experimental absorption spectrum. In addition, BSE calculations of the Te $M_{5}$ pre-edge transitions are performed using the OCEAN software package. The resulting OCEAN-simulated spectrum is plotted in Fig. 3(b), after applying a global shift to the energy scale to match the onset of the

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**FIG. 3.** (a) Band structure of 2H-MoTe$_2$ along the $\Gamma$–$K$–$M$–$\Gamma$ path. One representative VB $\rightarrow$ CB transition induced by the 400 nm pump pulse is shown as a blue arrow. (b) Experimental static XAS, CB DOS, and OCEAN-simulated spectra are plotted and horizontal dashed lines are drawn at representative energies within the CB, labeled CB$_1$, CB$_2$, and CB$_3$. (c) A model of the probing scheme by x-ray transient absorption of 2H-MoTe$_2$. The VB and CB are shown schematically as filled and empty boxes and photoexcited electrons and holes are drawn as filled or empty circles, respectively. Representative x-ray transitions from the Te 3d$_{5/2}$ core level are shown with and without an X to represent a decrease or increase in absorption relative to the static spectrum, respectively, due to state filling.
Capturing changes in the electronic structure with x-ray transient absorption

In the femtosecond time-resolved experiments performed at the PAL-XFEL, we photoexcite 2H-MoTe2 with a sub-100 fs, 400 nm pulse and measure the time-delayed XAS of the excited sample. The probing scheme is schematically depicted in Fig. 3(c). In the leftmost panel labeled \( t = 0 \) fs, the above-bandgap pump excitation (\( h\nu/400\text{ nm} = 3.1\text{ eV} \)) produces nonthermal excited carriers with an energy distribution centered significantly above the CB minimum (electrons) and below the VB maximum (holes). The state-filling signal is expected to map this distribution: the Mo M4,5 edge could also be used, in principle, to map changes in VB/CB state occupation. However, as seen in Figs. S5 and S6 of the supplementary material, the Mo M4,5 edge is quite broad, leading to a mostly featureless pre-edge spectrum. The Mo M4,5 edge is also weak compared to the overlapping N K edge, which hinders the measurement in this region due to the use of Si3N4 as a substrate as well as a target for the I0 measurement.

Femtosecond transient x-ray spectroscopy of 2H-MoTe2

In Fig. 4(a), the Te M5-edge absorbance of 2H-MoTe2 measured at PAL-XFEL without photoexcitation is plotted (ground state, blue line) and at a delay time of \(-400\) fs after optical excitation (excited state, orange line). The excited-state spectrum is an average of two transient spectra measured at nominal delay times of \(+300\) fs and \(+500\) fs (relative to the independent determination of time-zero described in the Experimental Methods and Calculations section). In the lower panel of Fig. 4(a), the differential absorption spectrum at 400 fs delay, defined as \( \Delta OD(E, \tau) = OD_{400\text{ fs}}(E, \tau) - OD_{\text{static}}(E) \), is plotted. This differential spectrum consists of three main features. First, an increased absorption at 572.5 eV, approximately 1 eV below the 3d5/2 \( \rightarrow \) CB onset. Second, a derivative feature centered at the 3d5/2 \( \rightarrow \) CB edge onset, with increased absorption just below the edge and decreased absorption just above the edge. Finally, it also shows a general decrease in the absorption at energies above the 3d5/2 \( \rightarrow \) CB edge (between 574 eV and 576 eV). The observed changes in the absorption amplitude are about \( \pm 7\% \) of the Mo pre-edge feature (static 3d5/2 \( \rightarrow \) CB absorption). Thus, the expected optically induced transient changes in the transmission are approximately 0.14% (i.e., 7% of the \( \pm 2\% \) change in transmission in the static pre-edge absorption). Due to the small absorption edge jump at the Te M5 pre-edge, the difference spectrum in Fig. 4(a) required averaging over 10 h of data.

The features observed in the difference spectrum at \(-400\) fs delay time are interpreted in terms of the middle panel of the schematic shown in Fig. 3(c). The increased absorption at 572.5 eV in the transient spectrum, \(-1\) eV below the 3d5/2 \( \rightarrow \) CB onset, is assigned to the photoexcited holes in the valence band. The derivative feature centered at 573.5 eV as well as the decrease in absorption above 574 eV are caused by a combination of bandgap renormalization (red-shifting of the edge), broadening due to the presence of free carriers and lattice-heat effects, and state filling by the electrons in the CB.

In Figs. 4(b)–4(d), \( \Delta OD \) measured at three chosen energies indicated by arrows in Fig. 4(a) are plotted as a function of time delay. The three chosen energies are (i) 572.9 eV, which corresponds to the 3d5/2 \( \rightarrow \) VB hole signal close to the VB maximum, (ii) 573.9 eV, which corresponds to the 3d5/2 \( \rightarrow \) CB signal close to the bottom of the CB (near the K and M critical points), and (iii) 576 eV, which corresponds to the 3d5/2 \( \rightarrow \) CB signal involving higher-energy states within the CB (near the \( \Gamma \) critical point). The features at 572.9 eV (3d5/2 \( \rightarrow \) VB) and 576 eV (3d5/2 \( \rightarrow \) CB) both appear to rise within the pump excitation at time-zero, whereas the feature at 573.9 eV (3d5/2 \( \rightarrow \) CB) appears to be slightly delayed. The rise of the 576 eV signal is fit to a convolution between a Gaussian function and a step function and the instrument-response function (IRF) is determined to be \( \pm 200\) fs, defined by the full width at half maximum (FWHM) of the Gaussian function. The time traces of the signals at 572.9 eV and 573.9 eV are each fit to a Gaussian broadened step function convolved with an exponential decay function. The onset of the 573.9 eV feature, defined by the center of the Gaussian broadened step function, \( t_0 \) is extracted as \( t_0 = 400 \pm 90\) fs and this is compared to the fitted onset of the 576 eV signal at \( t_0 = 0 \pm 70\) fs. The uncertainties in \( t_0 \) are the standard errors of the extracted best-fit parameter. The total delay in the onset of the 573.9 eV feature is therefore \( \Delta t = 400 \pm 110\) fs with respect to the fitted onset of the 576 eV signal, with the corresponding uncertainty determined after error propagation.
The appearance of the $3d_{5/2} \rightarrow VB$ and $3d_{5/2} \rightarrow CB_3$ signals at time-zero, within the IRF of the measurement, is consistent with a state-filling description where a distribution of carriers is produced immediately with the pump excitation. The optical excitation opens new transitions from the Te $3d_{5/2}$ core level to the transient holes in the VB ($3d_{5/2} \rightarrow VB$) and leads to a state-filling signal by the photoexcited electrons reaching energies up to 2.2 eV above the CB minimum ($3d_{5/2} \rightarrow CB_3$). The $\Delta t = 400 \pm 110$ fs feature near the band edge at 573.9 eV ($3d_{5/2} \rightarrow CB_1$), on the other hand, is attributed to the time required for the electrons to cool to the bottom of the CB by intra- and intervalley scattering through phonon emission, as depicted schematically in Fig. 3(c) (left to middle panels). The electron–phonon scattering leads to an electron distribution closer to the CB minimum and to a delayed state-filling response at these $3d_{5/2} \rightarrow CB_1$ energies, which is expected on the few-hundred femtosecond timescale. The measured electron cooling time of $400 \pm 110$ fs assigned here for 2H-MoTe$_2$, following above-bandgap photoexcitation, is comparable to the hole cooling time of $380 \pm 90$ fs measured in the same material by XUV transient absorption. The $400 \pm 110$ fs delay measured in the present work between state-filling signals at CB$_1$ compared to CB$_3$, which we attribute here to electron cooling, was not visible in the XUV measurements due to overlapping signals of the Te 4d$_{5/2}$ and 4d$_{3/2}$ core levels. The 1.5 eV spin–orbit splitting of the Te N$_{5,4}$ edges in the XUV prevents the distinction between CB$_3$ and CB$_1$, even in the static absorption measurement. We note that the 400 nm excitation in the present work produces carriers with considerably greater above-bandgap energy per photon compared to the broadband visible excitation (550–950 nm) used in the XUV experiment.

After the initial rise of the transients observed in Figs. 4(b)–4(d), the $3d_{5/2} \rightarrow CB_3$ feature at 576 eV remains constant up to a delay time of 4 ps, whereas the time traces of the $3d_{5/2} \rightarrow VB$ and $3d_{5/2} \rightarrow CB_1$ features are characterized by complete or partial decay on this timescale.
fit to a single exponential function $y(t) = y_0 + A \exp(-t/\tau)$, where $y_0$ and $A$ are arbitrary constants) and the extracted time constant, $\tau$, is found to be $\tau_{\text{hole}} = 1.2 \pm 0.6$ ps for the $3d_{x^2-y^2}$ to VB hole signal and $\tau_{\text{electron}} = 1.2 \pm 0.8$ ps for the $3d_{x^2-y^2}$ to CB$_1$ feature. These observations are consistent with electron–hole recombination and trapping, which occur on these timescales, i.e., leading to decay of the transient holes in the VB and of the photoexcited electrons in the CB. Although the carrier dynamics are expected to be dominated by an Auger recombination process, the exponential decay fitting is chosen to compare to exponential time constants extracted from recent THz and visible pump–probe studies on similar defect-rich 2H-MoTe$_2$ samples. Our measured electron and hole lifetimes ($\tau_{\text{hole}}$ and $\tau_{\text{electron}}$) are consistent, within the error bars, with the carrier-averaged lifetimes extracted from the THz and visible pump–probe studies and the hole lifetime measured by XUV transient absorption.

In addition to the consideration of state-filling effects, carrier–phonon scattering leads to lattice heating, which can also affect the $3d_{x^2-y^2}$ to CB absorption of 2H-MoTe$_2$, even in the absence of photoexcited carriers. The observation of a long-lived, negative plateau in the $3d_{x^2-y^2}$ to CB OOD signals at 573.9 eV and 576 eV after ~1 ps is associated with these lattice heating effects, which dominate after carrier cooling and electron–hole recombination. To corroborate this assignment, the analogous observation of this effect was made at the Te $N_5$ edge (Te 4f$^2_{5/2}$ to CB) of 2H-MoTe$_2$ using XUV transient absorption following both 800 nm and 400 nm photoexcitation (see the supplementary material for details). Similar long-lived, heat-induced effects are common in transient core-level absorption experiments of semiconductors.

CONCLUSIONS AND OUTLOOK

We measured ultrafast soft x-ray absorption spectra of polycrystalline 2H-MoTe$_2$ thin films following 400 nm excitation using an XFEL at PAL. The differential absorption spectrum at a delay time of 400 fs provides a direct observation of photoexcited holes in the VB, which are identified by short-lived Te $3d_{x^2-y^2}$ to VB absorption transitions appearing ~1 eV below the $3d_{x^2-y^2}$ to CB absorption edge. The transient spectrum is also characterized by a negative OOD signal at the $3d_{x^2-y^2}$ to CB energies due to excited electrons in the CB, bandgap renormalization, and lattice-heating effects. The time traces of the Te $3d_{x^2-y^2}$ to VB and Te $3d_{x^2-y^2}$ to CB$_1$ state-filling signals reveal hole and electron lifetimes of $1.2 \pm 0.6$ ps and $1.2 \pm 0.8$ ps, respectively, which is consistent with electron–hole recombination occurring on this timescale. These carrier lifetimes are also consistent with THz and optical pump–probe measurements, which previously provided carrier-averaged lifetimes. Prior to electron–hole recombination, we observe a $400 \pm 110$ fs delay of the $3d_{x^2-y^2}$ to CB$_1$ state-filling depletion involving states near the bottom of the CB compared to the $3d_{x^2-y^2}$ to CB$_3$ depletion involving higher-energy states within the CB. We interpret this finding as a direct signature of hot electron relaxation within the CB due to intra- and intervalley scattering by electron–phonon interactions. The observation of electron cooling dynamics in the present work provides additional and complementary information to the recent XUV transient absorption study of 2H-MoTe$_2$, which did not distinguish these electron relaxation dynamics due to overlapping spectral features of the neighboring spin–orbit split $N_5$ and $N_4$ edges.

HHG-based XUV transient absorption spectroscopy has a number of unique capabilities including: (1) demonstrated attosecond temporal resolution in the condensed phase, (2) simultaneous absorption at multiple elemental edges in a single measurement, and (3) intrinsic pump–probe delay stability due to the single amplified laser source for the pump and probe. The results presented here provide a benchmark for femtosecond transient absorption spectroscopy of semiconductor materials using the advancing capabilities of XFELs that operate at higher energies in the soft x-ray regime. Some of the distinct, promising, yet complementary capabilities offered by XFELs include: (1) access to higher-energy core edges with large spin–orbit splittings (>10 eV), which was exploited in the present work to minimize the adverse overlap of adjacent core-level absorption, (2) much higher photon flux per pulse, allowing for increased sensitivity via pulse-to-pulse normalization, surface sensitivity via TEY measurements, and the possibility to detect weak transient signals due to the reduction of the shot noise level, (3) smaller focal spot sizes (<1–10 μm), and longer absorption lengths at higher-energy x-rays, enabling future study of small semiconductor samples or individual domains of polycrystalline samples and thicker samples, respectively, and (4) greater tunability over the soft, tender, and hard x-ray energies for accessing K, L, M, etc., edges of elements, which opens the possibility of probing different orbital symmetries of the valence shell due to dipole selection rules. High-repetition rate XFELs, such as the European XFEL and LCLS-II (currently under construction) will further advance the sensitivity of this approach.

AUTHORS’ CONTRIBUTIONS

A.B. and A.R.A. contributed equally to this work.

SUPPLEMENTARY MATERIAL

See the supplementary material for details on the sample preparation and characterization, excited carrier density estimation, PAL-XFEL energy calibration, partial/projected density of states calculations, static Mo M3-edge measurements, and XUV transient absorption results with 400 nm pump–long lived signals.

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DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.
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