The Effect of Synthesis Temperature on Carbon Nanospheres from Palm Kernel Fiber

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Abstract
Carbon nanospheres (CNS) were synthesized successfully from palm kernel fiber activated carbon. Palm kernel fiber (PKF) which is an agro-waste was carbonized followed by physical activation with CO₂, then go along with treated utilizing ethanol vapor at temperature of 700 °C, 850 °C and 1000 °C. The temperature effect on developed synthesized nanomaterials was investigated using scanning Fourier transform infrared microscopy (FTIR), electron microscopy (SEM), energy dispersive x-ray spectroscopy (EDX), transmission electron microscopy (TEM), x-ray diffraction (XRD), and thermo-gravimetric analysis (TGA). The temperatures were varied from 700 °C to 1000 °C with intervals of 150 °C. The SEM results showed highest purity and the largest number of carbon-nanospheres being formed at a synthesis temperature of 1000 °C. The results indicate that CNS diameter, growth rate, crystallinity and density can be affected by increase in temperature. The CNS diameters were found to increase when the synthesis temperature increased. The results of the TEM showed that, within the temperature range of 700 °C to 1000 °C, the CNSs diameter increased continuously from 3 nm to 65 nm. XRD analysis showed that the synthesized carbon nanomaterials were amorphous. In general, the results showed that the synthesis temperature affects the diameter, density and crystallinity of carbon nanomaterials. The synthesis temperature of 1000 °C appears to be the ideal temperature for high quality and high yield CNSs production. The presence of iron (Fe) in the EDX results showed that such synthesized CNSs can be used as reinforcement materials in the manufacture of automobile brake pads.

Keywords: Carbon nanosphere, characterization, bio-based precursors, temperature

I. INTRODUCTION
The increasing need for more environmentally friendly and biodegradable material requires the development of nanomaterials.

Since Iijima's innovation of multi-walled carbon nanotubes in 1991, global enthusiasm in academic and industrial circles for carbon nanotubes (CNTs) and their distinctive characteristics has been high because of their nanoscale structures [1] Carbon-based nanostructures with various morphologies have created great interest in scientific research due to their outstanding and distinctive characteristics. The attention of scientists around the world has been drawn to research on new material technology, development and improvement of materials properties and the discovery of alternative precursors that impart the desired properties to the materials.

Chemical vapor deposition (CVD) [2, 3], laser ablation [4], and arc discharge [5] are the commonly used synthesis methods to produce CNTs. Among the variety of synthetic methods, CVD is the most effective technique for large-scale CNTs [6, 7]. The capability to grow nanotubes exactly on the nanomaterials by using the CVD method allows sufficient control for production of high purity nanotubes. The quantity of carbon that enables better CNT nucleation is reduced as particle size decreases [8-10].

Since the innovation of carbon nanotubes, carbon nanoparticles carbon nanofibers, and carbon nanostructures are becoming commercially significant and their importance has grown rapidly over the past decade. CNTs have been widely examined due to their unique mechanical, magnetic and electronic properties [11-13]. The CNTs are regarded as possible filler materials to enhance the mechanical properties as well as physical properties of polymer composites [14]. CNT applications are eco-friendly and offer good opportunities as well as new technology for various sectors which include biotechnology, aerospace, automotive and electronics industries.

By using ethanol as the carbon resource in the CVD method has positive impacts for instance, low temperature reaction, high product purity and the potential for reduced cost for large-scale production [15, 16]. Generally, the temperature of synthesis is a factor affecting crystallinity [17], density [18], diameter, growth rate [17-20], yield, morphology [21, 22] and purity [23] of nanomaterials. Investigations of temperature effect on CNT growth parameters have shown that the temperature range from 800 °C to 1000 °C is the optimal temperature which results in the highest yield of CNTs [23, 24]. When the temperature increases from 750 °C to 950 °C the
diameter, crystallinity, growth rate and density of CNTs can be measured and high levels of crystalline perfection can be achieved at temperature of 950 °C [18]. The interchange of temperature from 700 °C to 830 °C has been shown to result in structural defects in CNTs and graphite sheets grown [25].

Muataz et al. [20] varied the temperatures reaction of CNT from 500 °C to 850 °C and the result indicated that CNT growth was seen from 600 °C and high purity was obtained at a temperature of 850 °C. The study by Lee et al. [19] showed that the thicker layers of the compartment appear at 1100 °C more frequently. When the temperature gradually increased from 850 °C to 1100 °C, the comparative amount of crystalline graphitic sheets also increased. Toussi et al. [26] showed that a small amount of CNTs were formed at temperatures below 700 °C, while more shapeless carbons were formed in the CNTs at temperatures above 900 °C. Their study found that the optimum temperature for CNT growth was between 800 °C and 900 °C and the ideal temperature growth was at 850 °C. Pham et al. [27] reported that below the synthesis temperature of 800 °C, tube length growth and diameter was minimal, while the highest tube crystallinity was achieved at a temperature range between 800 °C and 840 °C. Madani et al. [17] presented the temperature effect variation on the growth of CNTs utilizing CVD and the results revealed that the CNTs diameters increased when the temperature of synthesis increased, but the CNTs growth crystallinity decreased.

Zhao et al. [28] revealed that the CNTs multiplied quicker as the temperature increased, while the catalyst showed a significant increase for a little while, followed by a decrease. As temperature increased from 805 °C to 830 °C, the crystallinity improved a little while the height and the yield of CNT forests tripled in the course of increasing from 800 °C to 1100 °C. Shamsudin et al. [24] achieved 99.99 % optimum growth at a temperature of 900 °C. According to Jiang and Lan [29], at a low temperature of 450 °C, CNTs can be synthesized from carbonaceous solids, as opposed to high temperature synthesis. This paper presents the result of the temperature effect on synthesis of carbon nanomaterials at 700 °C, 850 °C and 1000 °C for 30 min treated with ethanol vapor. In the manufacturing of CNTs, nanospheres and nanosheets, the synthesis temperature performs a significant role. Generally, the temperature of synthesis is a factor that affects the final purity, morphology and yield of a nanomaterial.

III. MATERIAL CHARACTERIZATION

The morphology of the synthesized CNS was investigated utilizing FEGSEM (Zeiss, Germany Model: Ultra Plus) with an EDX in-built which was used for elemental composition analysis. The sample phase identification and microstructure were examined using JEOL 1400 TEM and a JEOL HRTEM 2100 model at a speed voltage of 20 kV. The sample of CNS needed for HRTEM measurements was approximately 0.0002 g mixed with ethanol using an 80 Hz Watt transistorized sonic cleaner for 20 min. The TEM preparation of the carbon nanosphere mixed with ethanol was achieved by putting a droplet of the solution on formvar with painted 150 meshes of copper grids. The TEM sample was dry at lower temperature and watched at 100 kV speed voltage.

For TEM, the iTEM software and Gatan camera were used to store and process the images taken digitally from the camera (Megaview III). The backloading preparation method was used for the preparation of XRD samples of the PKF nanomaterial. The Netherlands made PANalytical Empyrean Diffractometer with a Co-Kα radiation and X’Celerator detector was used to analyze the sample. The generator configuration was 40 kV and 40 mA. A PerkinElmer Product was used to study the FTIR of the sample. A PerkinElmer synchronized thermal analyzer (STA) 6000 was used to establish the thermal stability of the synthesized nanosphere material.

IV. RESULTS AND DISCUSSION

A. SEM

Increasing the temperature of the furnace can increase the frequency of impact of the elements found in carbon activated PKF resulting in various structures being formed as shown in Fig. 1.
Fig. 1 (b-d) shows the SEM images of CNS produced from PKF at different buildup temperatures 700 °C, 850 °C and 1000 °C. FEGSEM and HRTEM were used to characterize the temperature effect on the synthesized nanomaterials developed. Fig. 1a shows the SEM image of physically activated PKF produced at 850 °C for 60 min under CO₂. The micrographs SEM image from Fig. 1b to d present carbon sheets, carbon-nanoparticles (CNPs) and carbon-nanospheres (CNSs). The SEM image at 700 °C from Fig. 1b indicates the presence of tiny sphere particles or sheet-like structure grown with irregular morphology. Low density nanospheres with small diameter ranging from 3 nm to 20 nm were formed. This indicates that the synthesis temperature of 700 °C was not enough to produced carbon nanospheres/nanotubes. At the temperature of 850°C, a small sphere-like morphology was noticed as shown in Fig. 1c. At this temperature several carbon nanospheres with small quantities of amorphous carbon were developed.

The results showed a huge increase in hydrocarbon pyrolysis with particle distribution size ranging 10 nm to 65 nm when the synthesis temperature increased from 850 °C to 1000 °C, as shown in Figure 1d. Increasing the temperature to 1000 °C produced clean nanospheres with aligned and uniform diameter distribution size. The SEM micrographs of all developed
samples display a strong difference in the structure of the synthesized materials at the different temperatures. Thus, the morphological analysis under atmospheric pressure showed that the temperature build-up played an important role in the synthesis of uniform CNSs.

B. Elemental Composition Analysis

A FEGSEM built-in with EDX was employed to analyze the activated carbon element composition and synthesized nanomaterials at different temperatures. Table I shows the PKF activated carbon EDX analysis and synthesized nanomaterials at temperatures of 700 °C, 850 °C and 1000 °C. From the table there is a high carbon content in all samples with the highest yield for all synthesis temperatures from nanomaterials synthesized at 850 °C. After the carbonized PKF was physically activated at 850 °C under CO₂, it was followed by treatment for 30 min with ethanol vapor to produce carbon nanospheres. This temperature has the highest carbon nanomaterials yield because the synthesis temperature is high enough to produce carbon nanospheres while also decreasing amorphous carbon. In addition, EDX showed a high carbon content of 88.10 %, 90.06 % and 89.40 % respectively in all synthesized carbon nanosphere samples; this confirmed the treatment of the nanomaterials with ethanol vapor. From Table I it is evident that as the synthesis temperature increased the oxygen content reduced which agrees with previous results obtained [30].

| Table I. EDX analysis of PKF carbon activated, and nanomaterials at the different temperature |
|-----------------------------------|---------------------------------|---------------------------------|---------------------------------|---------------------------------|
| Element      | Activated Carbon Wt.% | Nanomaterial Synthesized at 700°C Wt.% | Nanomaterial Synthesized at 850°C Wt.% | Nanomaterial Synthesized at 1000°C Wt.% |
|--------------|------------------------|----------------------------------|----------------------------------|-----------------------------------|
| C            | 87.59                  | 88.10                            | 90.06                            | 89.40                             |
| O            | 7.80                   | 6.85                             | 6.76                             | 6.52                              |
| Mg           | 0.79                   | 0.76                             | 0.54                             | 1.27                              |
| Al           | 0.32                   | 0.97                             | 0.52                             | 1.17                              |
| Si           | 0.97                   | 1.29                             | 0.57                             | -                                 |
| K            | 1.09                   | 1.53                             | 1.20                             | 1.12                              |
| Fe           | 0.23                   | 0.50                             | 0.34                             | 0.52                              |
| Cu           | 1.20                   | -                                | -                                | -                                 |
| Total        | 100.00                 | 100.00                           | 100.00                           | 100.00                            |

The elemental composition in PKF carbon activated for CF included oxygen (O), carbon (C), and potassium (K), which are normally originated from plants. The existence of oxygen in all samples developed can be credited to the carbonized PKF physically activated under CO₂ and ethanol aerosol. The existence of silicon (Si) was noticed in the synthesized nanomaterials which can be attributed to the reaction of the material in the quartz tube. The results from both FESEM and EDX established that the samples produced form PKF activated carbon are carbon nanospheres. The presence of iron (Fe) in all the samples as seen in Table I showed that the nanomaterial can be used as additive materials in the production of brake pads.

C. High-Resolution TEM (HRTEM)

The TEM analysis was conducted using a JEOL HRTEM 2100 model to study the morphological changes and the material deflections and the actual transition that appears in the carbon nanomaterial structure. Fig. 2(c) shows small quantities of CNSs at 850 °C with particle size distribution as seen in the TEM image varying from 10 nm to 40 nm diameter; however, the sphere structure was not clearly identified. At this temperature, increased numbers of CNSs with a smaller amount of deformation and a higher CNS yield was observed with fewer impurities.

As seen in Fig. 2(c), the TEM images show uniform nanosphere sizes, while some nanospheres have some deformation and are very weak. The TEM at 1000 °C as seen in Figure 2 (d) shows that high purity and high yield CNSs were formed with high agglomeration. The distribution size diameter of nanospheres range from 10 nm to 65 nm, while nanospheres with 25 nm to 35 nm diameter occur at a higher frequency than those of 40 nm to 65 nm diameter.
At 700 °C, the absence of carbon nanomaterial growth indicates that the temperature at that level was not adequate high to transform the PKF carbon activated into tubes or spheres. As ethanol vapor was added and the temperature increased, the state of PKF activated carbon content in the tube was modified, thereby activating the carbon nanospheres growth on carbon activated. It was noticed that temperature of 1000 °C is appropriate to develop nanospheres.

D. XRD Analysis

The structural characterization of carbon nanomaterial samples at different synthesis temperatures was achieved using XRD technique using the Netherlands made PANalytical Empyrean Diffractometer including Co-Kα radiation and X'Celerator detector. Fig. 3 is the XRD pattern obtained from activated carbon at 850 °C for 60 min and synthesized nanomaterials at temperature for 30 min. All the synthesized nanomaterials result at 700 °C, 850 °C to 1000 °C displayed two major broad diffraction peaks in Fig. 3 (b-d). The 2θ value (002) diffraction peak was found at 22.11°, whereas the (101) diffraction peak was discovered at 42.96°, indicating hexagonal graphite structural planes.
The peak located at (002) confirmed the presence of lignin and hemicellulose contents found in PKF and all the samples were amorphous in nature. These two peak values exhibit low crystallinity and high amorphous carbon structure of the nanomaterials sample and are in accordance with results from other studies [23, 29].

E. SAED

The HRTEM was utilized to establish the effect of temperature on crystallinity of nanomaterials. Fig. 4 shows the image of the SAED and HRTEM rings of the nanospheres produced in the research. From Fig. 4(a) and (c) it is evident that the CNSs have a strongly regular crystalline structure. The HRTEM reveals a small portion of nanosphere grown at temperature of 850 °C and a graphite structure near the edge of the sphere as shown in Fig. 4(a). Increasing in temperature from 850 °C to 1000 °C, the HRTEM image of CNS showed that most CNSs were formed at unique depths with sphere curves as shown in Fig. 4(c). The SAED pattern has wide diffraction rings that show the presence of nanosphere crystallinity and graphitic nature which agrees with the planes (002) and (101) and displays agreement with the results from XRD.
The rings brighter areas in Fig. 4b and d agree with the hexagonal graphite reflection 002, whereas the next constant ring observed the diffraction pattern agrees to the hexagonal graphite reflection 101. The SAED pattern from CNS samples at 850 °C as shown in Fig. 4b show some sets of diffuse diffraction and unclear spots, showing that the material has a little crystalline or amorphous carbon, while the diffraction rings from Fig. 4d showed that the CNS samples at 1000 °C present amorphous carbon structures and crystalline together.

F. TGA

TGA was used to examine the thermal stability of the synthesized nanomaterial at different temperatures as shown in Fig. 5. The analysis was carried out under nitrogen gas (20 ml/min flow rate) and heating rate (10 °C/min) from 30 °C to 600 °C. The synthesized nanomaterials at 700 °C, 850 °C and 1000 °C began to vaporize at the temperature of approximately 200 °C, 350 °C and 430 °C respectively. The early weight loss for all the nanomaterials happened at temperatures within 100 °C to 150 °C as a result of the moisture content, as well as loss of the synthesized sample functional group and other non-carbon material present in the sample. The synthesized nanospheres at 700 °C began to decay at a temperature of about 200 °C (thermal decomposition), which was caused by the presence of amorphous carbon at a lower synthesis temperature.

![Fig. 4](image_url)  
Fig. 4 (a) HRTEM of CNS synthesized at 850°C (b) SAED at 850°C, (c) HRTEM of CNS synthesized at 1000 °C and (d) SAED at 1000°C

![Fig. 5](image_url)  
Fig. 5 TGA results for Synthesized nanomaterials at (b) 700 °C (c) 850 °C and (d) 1000 °C
The initial degradation temperature for synthesized nanospheres prepared at 850 °C started at 350 °C due to less amorphous carbon produced by longer reaction times, which led to production of better nanospheres. At 1000 °C, the synthesis temperature was high enough that almost no amorphous carbon was burned. The nanomaterial synthesized at 1000 °C started degrading at 430 °C which agrees with the decomposition of carbonaceous materials functional groups, while the whole weight loss for synthesized nanomaterials took place at 560°C. This indicates that small diameter nanospheres were formed at low temperature and nanosphere diameter increased when the temperature of the synthesis increased.

In conclusion, the synthesized nanospheres thermal stability improved at 1000 °C. The TGA results agreed with the EDX and SEM results and showed that within this temperature range, functional oxygen groups started to release, causing aromatization activation. The results also provide clear evidence that the crystalline perfection degree of CNSs increases with increasing temperature.

G. FTIR

The FTIR was measured between 400 cm⁻¹ to 4000 cm⁻¹ to characterize the synthesized nanomaterial functional groups using Perkin Elmer Product

Fig. 6 shows the FTIR spectrum of activated carbon PKF and synthesized nanomaterials from PKF at 700 °C, 850 °C and 1000 °C for 30 min. From the FTIR result, all the PKF synthesized nanomaterials showed strong peaks at 1071 cm⁻¹ to 1220 cm⁻¹ which indicates different characteristics of lignin composition (O-H bending vibrations and C-O stretching) groups, which reduced as the temperature increased.

The FTIR spectra of all synthesized nanomaterial samples show similar vibrational modes as well as different strengths and energy changes. As shown in Fig. 6 (b) and (c), the FTIR at 700 °C and 850 °C reveals plane peaks of bending C-H compounds between 693 cm⁻¹ and 776 cm⁻¹, signifying the carbon bend inside the aromatic ring. When the synthesis temperature increased 850 °C to 1000 °C, the FTIR spectrum shows that there are no other peaks which indicates that the nanomaterial product has high purity.

V. CONCLUSION

Synthesized carbon nanospheres were successfully produced using ethanol vapor through activated carbon made from palm kernel fiber at temperatures of 700 °C to 1000 °C synthesis with 150 °C increase. The SEM analysis results showed highest purity and number of carbon-nanospheres formed at 1000 °C. The results indicate that CNS diameter, growth rate, crystallinity and density can be affected by increase in temperature. Also, the temperature played a major role in the production of high quality and high yield CNSs. We noticed that at 1000 °C, the carbon nanospheres (CNSs) were formed from synthesized nanomaterials of different sizes and shapes. The SEM images showed that the temperature growth affected the CNSs growth rate as well as their diameter and density. As the temperature increased, the increase in growth rate of synthesized CNSs was most likely due to the increased carbon diffusion and reaction rate. It was also discovered that the temperature was an influential factor in regulating the structure of the CNSs.
In conclusion, synthesis temperature greatly affects the diameter, density and crystallinity of carbon nanomaterials synthesized from PKF. The synthesis temperature of 1000 °C can be considered as the ideal temperature for production of high quality and high yield CNSs. The EDX result showed the presence of iron (Fe) in the sample which makes PKF nanomaterials good candidates as additive material in the production of brake pads.

NOMENCLATURE

CNS  Carbon Nanospheres
PKF  Palm Kernel Fiber
CF   Carbon Fiber
O    Oxygen
C    Carbon
K    Potassium
CO₂  Carbon dioxide
Si   Silicon
Fe   Iron
SEM  Scanning Electron Microscope
XRD  X-Ray Diffraction
EDX  Energy Dispersive X-Ray
CVD  Chemical vapor deposition
TGA  Thermo-gravimetric analysis
SAED Selected Area Electron Diffraction
CNTs Carbon nanotubes
CNPs Carbon-Nanoparticles
FEGSEM Field Emission Gun Scanning Electron Microscope
FTIR Spectroscopy Fourier Transform Infrared Microscopy
TEM Transmission Electron Microscopy
HRTEM High-Resolution Transmission Electron Microscopy

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