Effect of ball milling times on phase formation of Mg$_{0.7}$Zn$_{0.3}$Fe$_2$O$_4$

Aurawan Rittidech and Panuthat Khotsongkram
Department of Physics, Faculty of Science, Mahasarakham University, Mahasarakham, Thailand

Abstract: Mg$_{0.7}$Zn$_{0.3}$Fe$_2$O$_4$ (MZF) powder was prepared by a solid-state mixed oxide using ball-milling technique. The phase identification of calcined powders was performed using X-ray diffraction technique (XRD). It has been found that MZF powders were successfully obtained for calcination conditions of 800 °C for milling times of 36 h, whereas milling times of 12 h, unreacted ZnO, Fe$_2$O$_3$ and MgO phases (2θ ~ 54, 57 and 63) are detected from the original mixture up to 900 °C. By increasing the calcination temperature to 1000 °C, a single phase of MZF is formed. Moreover, the microstructure of powder were examined by scanning electron microscopy (SEM) and showed that the particle in size from 58-1200 nm. While, XRD patterns were obtained average size of particle, calculation by Scherrer’s equation, between 30.54 – 66.59 nm.

Keywords: Mg$_{0.7}$Zn$_{0.3}$Fe$_2$O$_4$, ball-milling, calcination

INTRODUCTION

Magnetic ferrites were studied extensively due to their many applications. A candidate system of such a revisit is the mixed ferrites of Zn and Mg as magnesium-zinc ferrites (Mg$_{1-x}$Zn$_x$Fe$_2$O$_4$). Mg$_{1-x}$Zn$_x$Fe$_2$O$_4$ is one of a ferrites material, displaying interesting magnetic properties. The importance of ferrites is well realized in the fields of electronics and communication which has a wide range of magnetic coils and electronic devices [1]. It is found that magnesium-zinc ferrites fraction x between 0.2 to 0.3 were significantly obtained magnetic properties [2-3]. The crystal structure of these materials controls their properties. Then the magnetic properties of materials are sensitive to conditions of preparations. The preparations of materials by solid-state reaction route under ball milling technique are not complicate and inexpensive so they are suitable for mass product. Recently, many studies [3] indicated that X-ray diffraction were carried out for Mg-ferrites, Ni- ferrites, Ni-Zn ferrites, Ni-Mg ferrites, Mg-Zn ferrites, Ni-Zn-Mg ferrites.

The aim of this work is to study the effect of ball milling times on phase formation and microstructural development of Mg$_{0.7}$Zn$_{0.3}$Fe$_2$O$_4$ powders in order to find an optimum calcinations condition.

EXPERIMENTAL PROCEDURE

Mg$_{1-x}$Zn$_x$Fe$_2$O$_4$ powder with x = 0.3 was prepared from laboratory-grade oxides of zinc oxide (ZnO), ferric oxide (Fe$_2$O$_3$) and magnesium oxide (MgO). All raw materials have been prepared by a solid-state mixed oxide technique. In the mixing process, the calculated relevant proportions of constituents were weighed, suspended in isopropyl alcohol (IPA) and intimately mixed for different times in a ball-milling, for 12, 18, 24, 30 and 36 hours with metal media. Drying was carried out for 2 h (the sample dish was placed on a hotplate with a magnetic stirring in action to prevent gravitational separation of the components). The dried powder was then ground, sieved and calcined at temperature between 800-1200 °C. XRD technique (XRD; Philips PW 1729 diffractometer) was used to identify the optimum firing temperatures, for phase identification, for quantitative analysis of mixtures of phases. The means crystallite size D was obtained from the Scherrer formula [4]:

\[
D = \frac{(k\lambda)}{(\beta \cos \theta)}
\]

where \(k = 0.9\), \(\lambda = 1.541\), \(\theta\) is the diffraction peak angle, and \(\beta\) denotes the full width at half maximum of the corresponding diffraction peak. The particle sizes and morphologies of the powders have been directly imaged using scanning electron microscopes SEM and EDX (JEOL JSM-840A).

RESULTS AND DISCUSSION

All calcined powders together with that of the starting powder mixtures were examined by XRD in order to investigate the phase development (Fig.1 (a)-(f)).
As shown in Fig. 1(a), for non ball milling, XRD pattern of the uncalcined powder was showed peak of precursors ZnO, MgO and Fe$_3$O$_4$. By increasing the calcinations temperature from 800 to 1200 °C, the yield of the Mg$_{0.7}$Zn$_{0.3}$Fe$_2$O$_4$ phase increases significantly until at 1200 °C, single phase of MZF is formed. Having established the optimum calcinations temperature, powders with ball milling times at 12 h, 18 h, 24 h, 30 h and 36 h were calcined between at 800 and 1200 °C, as shown in Fig. 1 (b)-(f), respectively. In Fig. 1 (b)-(f), the single phase of MZF were found to be possible in 1000 °C for time of milling at 12 and 18 h, in 900 °C for time of milling at 24 and 30 h and in the last 800 °C for time of milling at 36 h. The optimal calcination conditions for phase of MZF is interested at two conditions, one at the short time of milling with 12 h which is calcined at 1000 °C and other at the long time of milling with 36 h which is calcined at 800 °C. This result was showed for the milling time which is optimized to fine powders; these can be matched with heat treatment for MZF phase formation.

Fig. 2: The relationship of average particles size from Scherrer’s equation and milling times with different calcination.

From XRD patterns were obtained average size of particle, calculation by Scherrer’s equation, between 30.54 – 66.59 nm and the relationship between particle size of all samples as a function of the calcination conditions and milling times are shown in Fig. 2. SEM shows the morphological evolution (Fig. 3), these result indicates that degree of grain size trend to increase with higher calcination temperature and the particles are decreased in shape with samples milled for longer time. The MZF particle size estimated by SEM micrographs as between 58-1200 nm and it is different in size by Scherrer’s equation since Scherrer’s equation is calculated from XRD patterns while SEM micrographs are calculated from selected area in sample. In general, similar morphological characteristics were observed in these powders, i.e. agglomerated and basically irregular in shape. EDX analysis and chemical
compositions for some MZF of these powders are shown in Fig. 4. As shown in Table 1 can be confirmed the parent of compositions to be MZF.

Fig. 3: SEM micrographs of MZF different conditions
(a) non ball milled with uncalcined
(b) milled for 12 h with calcined at 800 °C
(c) milled for 12 h with calcined at 1000 °C.

Table 1: Chemical compositions of the MZF powders from EDX analysis calcined at:
(a) 1000 °C with milling of 12 h
(b) 800 °C with milling of 36 h.

| Samples | Compositions (at%) |
|---------|-------------------|
|         | Fe(K) | Zn(K) | Mg(K) | O(K) |
| MZF (a) | 28.69  | 5.14  | 10.83 | 54.44 |
| MZF (b) | 34.57  | 5.95  | 8.89  | 49.82 |

Fig. 4 EDX analysis of the MZF powders milling for
(a) 12 h and calcined at 1000 °C
(b) 36 h and calcined at 800 °C.

SUMMARY

Single phase of MZF may be produced by solid state reaction via ball milling technique. The optimal calcination conditions for phase of MZF is obtained from two conditions, one at 1000 °C for milling time of 12 h and other at 800 °C for milling time of 36 h. The MZF particle size estimated by SEM micrographs as between 58-1200 nm, while from XRD patterns were obtained average size of particle, calculation by Scherrer’s equation, between 30.54 – 66.59 nm.

ACKNOWLEDGEMENTS

The authors express their sincere appreciation to Mahasarakham University, Thailand for the support.

REFERENCES

1. Rezlescu, N., E. Rezlescu, P.D. Popa, M.L. Craus and L. Rezlescu, 1998. Copper ions influence on the physical properties of a magnesium-zinc ferrite. J. Magnet. and Magn. Mater., 183: 199-206.
2. Chandrasekaran, G. and P. Nimy Sebastian, 1998. Magnetic study of Zn$_{x}$Mg$_{1-x}$Fe$_2$O$_4$ mixed ferrites. Mater. Lett., 37: 17-20.
3. Amer, M.A. and M. El Hiti, 2001. Mossbauer and X-ray studies for Ni$_{x}$Zn$_{1-x}$Mg$_{18-x}$Fe$_2$O$_4$ ferrites. J. Magnet. and Magn. Mater., 234: 118-125.
4. Rao, P., M. Iwasa, J. Wu, J. Ye and Y. Wang, 2004. Effect of Al$_2$O$_3$ additive on ZrO$_2$ phase composition in the Al$_2$O$_3$-ZrO$_2$ system. Ceram. Int., 30: 923-926.