Giant Rashba splitting in 2D organic-inorganic halide perovskites measured by transient spectroscopies

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Two-dimensional (2D) layered hybrid organic-inorganic halide perovskite semiconductors form natural “multiple quantum wells” that have strong spin-orbit coupling due to the heavy elements in their building blocks. This may lead to “Rashba splitting” close to the extrema in the electron bands. We have used a plethora of ultrafast transient, nonlinear optical spectroscopies and theoretical calculations to study the primary (excitons) and long-lived (free carriers) photoexcitations in thin films of 2D perovskite, namely, (C₆H₅C₂H₄NH₃)₂PbI₄. The density functional theory calculation shows the occurrence of Rashba splitting in the plane perpendicular to the 2D barrier. From the electroabsorption spectrum and photoinduced absorption spectra from excitons and free carriers, we obtain a giant Rashba splitting in this compound, with energy splitting of (40 ± 5) meV and Rashba parameter of (1.6 ± 0.1) eV·Å, which are among the highest Rashba splitting size parameters reported so far. This finding shows that 2D hybrid perovskites have great promise for potential applications in spintronics.

INTRODUCTION

The extrema points in the conduction band (CB) and valence band (VB) of semiconductors are of utmost importance in determining the optical, spin, and transport properties of these materials. The electron dispersion relation, \( E(k) \), near these extrema points is usually described by the effective-mass approximation, where the electrons and holes are treated as “free carriers” having an effective mass, \( m^* \), that leads to a spin-degenerate parabolic dispersion, \( E(k) = \hbar^2 k^2 / 2m^* \) (Fig. 1A). However, spin-orbit coupling (SOC) can split the spin-degenerate bands in noncentrosymmetric compounds, as first realized by Dresselhaus et al. (1) and Rashba (2). The effect of SOC may be enhanced in reduced dimensions, such as in two-dimensional (2D) semiconductors (3). In the presence of structural “inversion asymmetry,” the spin-degenerate parabolic band splits into two spin-polarized bands, where the electron (or/and hole) dispersion relation may be described by \( E_n(k) = (\hbar^2 k^2 / 2m^*) \pm \alpha_R |k| \), where \( \alpha_R \) is the Rashba splitting parameter. This formulation yields new extrema at a momentum offset (\( k_0 \)) and energy splitting (\( E_R \)) that are related to each other via \( \alpha_R = 2E_R/k_0 \) (Fig. 1B) (4). The two Rashba split branches have opposite spins, which can influence the photoexcitation’s optical and magnetic properties, a situation that may benefit spintronics because it enhances the spin-to-charge conversion efficiency (4, 5).

Large Rashba splitting with \( \alpha_R \) of few electron volt angstrom has been observed in only a handful of materials (such as ultrathin metallic films) (6), surfaces of topological insulators (such as Bi₂Se₃) (7), and surfaces of the polar semiconductor BiTeI (8). We note that Rashba splitting is very small in traditional III-V semiconductors. For example, the Rashba splitting parameter \( \alpha_R \) at the interface of InAlAs/InGaAs was measured to be \( \approx 0.07 \text{ eV·Å} \) with \( E_R \approx 1 \text{ meV} \) (9). In contrast, here, we discovered a giant-sized Rashba splitting in a 2D semiconducting layered hybrid organic-inorganic perovskite.

The 3D hybrid organic-inorganic perovskites (hereafter, hybrid perovskites), such as methyl ammonium lead halogen (MAPbX₃, where X is a halogen), have recently attracted immense attention because of several promising optoelectronic device applications, such as photovoltaic solar cells, light-emitting diodes, and lasers (10, 11). These compounds have strong SOC due to the heavy elements (Pb, X) that determine the electron bands near their extrema points, which may lead to large Rashba splitting if the structure lacks inversion symmetry (12); this is realized in some 2D and 3D hybrid perovskites (13). Interest in the hybrid perovskites for spintronics applications has only recently begun, with promising spin and magnetic field phenomena (14, 15); therefore, solid evidence of Rashba splitting existence in these compounds would further boost this interest.

Recently, the 2D hybrid perovskites have come into focus (16, 17). These compounds self-assemble into alternating organic and inorganic layers that form natural “multiple quantum wells,” with outstanding optoelectronic characteristics and stability under ambient conditions. Having strong SOC, high charge mobility, and intrinsic quantum well structures with many interfaces and facile solution processability, the 2D hybrid perovskites may be promising candidates for room temperature spintronic applications (18). Our work shows that these materials may also exhibit giant Rashba splitting, which could further enhance their appeal for this field.

RESULTS

We have used a variety of transient and steady-state linear and nonlinear optical spectroscopies, as well as theoretical calculations to study the optical characteristics of the primary (excitons) and long-lived (free carriers) photoexcitations in 2D hybrid perovskite films (see the Supplementary Materials). Figure 1C schematically shows the 2D hybrid perovskite that we studied here, namely, (C₆H₅C₂H₄NH₃)₂PbI₄ (PEPI), where the organic (C₆H₅C₂H₄NH₃)⁺ and inorganic [PbI₄]⁻ octahedron layers form multiple quantum wells with thickness of \( \approx 1 \text{ nm} \) (barrier) and \( \approx 0.6 \text{ nm} \) (well), respectively (19). A detailed description that includes energy gaps of the respective wells and barriers is given in fig. S1. The spectroscopies and model calculations used in this work, as...
Fig. 1. Introduction to Rashba splitting and the 2D layered hybrid perovskite PEPI. (A) Schematic electron dispersion relation of a regular CB that shows a doubly spin-degenerate parabolic band having a single minimum at \( k = 0 \). (B) Same as in (A) but subjected to Rashba splitting; two parabolic branches having opposite spin sense are formed. The Rashba energy \( E_R \) and momentum offset \( k_0 \) are denoted. (C) Structure of PEPI having alternating organic \( (\text{C}_6\text{H}_5\text{C}_2\text{H}_4\text{NH}_3)^+ \) and inorganic \( \text{PbI}_6^{2-} \) layers that form multiple quantum wells. (D) Absorption (Abs) and PL spectra of PEPI film at room temperature. a.u., arbitrary units.

detailed in Materials and Methods, provide compelling evidence that the continuum band edge above the exciton level (CB bottom and/or VB top) has surprising optical characteristic properties, which result from a large Rashba splitting energy, \( E_R \approx 40 \text{ meV} \). Our work provides a general all-optical method to study the Rashba splitting effect in semiconductors.

The room temperature photoluminescence (PL) and absorption spectra of the 2D hybrid perovskite PEPI film are dominated by an exciton band at \( \approx 2.4 \text{ eV} \) with large oscillator strength, consistent with a relatively large exciton binding energy, \( E_B \approx 0.2 \text{ eV} \) (Fig. 1D) (20), followed by a slow increase in the absorption with an onset at \( \approx 2.6 \text{ eV} \). However, at low temperatures (\( T < 110 \text{ K} \)), the absorption spectrum shows two step-like absorption edges in the spectral range of 2.45 to 2.65 eV (Fig. 2A and fig. S2). We thus used a modulation spectroscopy, namely, the electroabsorption (EA), to separate these delicate absorption features from the broad spectral background. Recall that the EA spectrum in neat semiconductors having intermediately large \( E_B \) usually contains two different spectral features: a Stark shift of the exciton below the continuum and a Frank-Keldysh (FK)–type oscillatory feature at the continuum band edge (21).

Figure 2B shows low-temperature EA spectra of the PEPI obtained at various ac field strengths. On the basis of the EA dependence on the field strength (or \( V \), the applied voltage), we identify two distinctive EA spectral ranges. The EA scales with \( V^2 \) (fig. S3) for \( h\omega > 2.55 \text{ eV} \) (22), but it saturates at large field for \( h\omega > 2.55 \text{ eV} \) (Fig. 2D, inset). The EA spectrum in the low-energy spectral region (\( < 2.55 \text{ eV} \)) shows a “first derivative–like” feature consistent with an exciton Stark shift, having a zero crossing at 2.38 eV that we assign as the 1s exciton energy, \( E_{1s} \) (Fig. 2F). We note that in this spectral region, there is a second derivative–like feature with “trending” zero crossing at \( \approx 2.53 \text{ eV} \), in agreement with the lowest step-like feature in the absorption spectrum (Fig. 2A). We identify this EA feature as originating from the 2s exciton in PEPI (23) instead of another exciton series (24), as shown in Fig. 2F (13, 25).

In contrast, the large oscillatory-like EA feature at \( h\omega > 2.55 \text{ eV} \) having multiple zero crossings is due to FK oscillation above the direct band edge (21, 26). The oscillation energy period \( \delta E \) shows “field broadening” that scales with \( V^{2/3} \) (Fig. 2C, to E). This broadening and the EA saturation at large \( V \) are typical characteristic properties of the FK oscillation in the EA spectrum close to the band edge [interband (IB) transition in Fig. 2B], where the peak at energy below the first zero crossing determines the energy gap value. We thus locate the band edge of the PEPI film at 2.57 eV (Fig. 2B). From \( E(B) \), \( E_{1s} \), and \( E_{2s} \) we can now obtain the 1s and 2s exciton binding energies, \( E_B(1s) = (190 \pm 4) \text{ meV} \) and \( E_B(2s) = (45 \pm 8) \text{ meV} \), respectively. The uncertainty originates mainly from the optical resolution of our spectrometer (2-nm entrance slit size) and the film inhomogeneity; all uncertainties are reported as 1 SD.

To further investigate the excited-state properties of PEPI close to the continuum band minima, we studied the primary photoexcitations using the picosecond transient photomodulation (PM) with 250-fs time resolution in a broad spectral range of 0.23 to 2.8 eV, excited at 3.1 eV (see Materials and Methods). The PM spectrum contains photoinduced absorption (PA) bands with \( \Delta T < 0 \) due to excited-state absorption and photoinduced bleaching (PB), with \( \Delta T > 0 \) caused by pump-induced bleaching of the ground-state absorption. At \( t = 0 \) ps, the PM spectrum contains two dominant spectral features (Fig. 3A): a PA band in the mid-infrared (IR) spectral range, denoted as PA1 at \( \approx 0.35 \text{ eV} \), and a PM feature in the visible spectral range centered at the PEPI exciton absorption (\( \approx 2.4 \text{ eV} \)) having both PA and PB components, as well as a weak PA band at \( \approx 1 \text{ eV} \) denoted as PA2. We note the occurrence of “zero sum” when adding the PA and PB bands of the entire PM spectrum, indicating that there are no other important photoinduced features in the spectrum. In addition, the PA and PM bands share similar decay dynamics (Fig. 3B), which supports the notion that they originate from a single primary photoexcitation species. We note that the PA dynamics contain a much slower component that is represented by a constant in the decay fitting (Fig. 3B), indicating that some long-lived photoexcitations survive the fast exciton recombination; these may be photocarriers, as discussed below.

The PM band in the visible range can be fit by linear combination of PB of the absorption spectrum and the absorption first and second features in the spectrum. In addition, the PA and PM bands share similar decay dynamics (Fig. 3B), which supports the notion that they originate from a single primary photoexcitation species. We note that the PA dynamics contain a much slower component that is represented by a constant in the decay fitting (Fig. 3B), indicating that some long-lived photoexcitations survive the fast exciton recombination; these may be photocarriers, as discussed below.

The PM band in the visible range can be fit by linear combination of PB of the absorption spectrum and the absorption first and second derivatives, as shown in Fig. S4. Therefore, it can be explained by a combination of “band filling” due to the primary photoexcitation (27), photoinduced symmetry-breaking process [such as spatially inhomogeneous strain or photoinduced electric field (18, 28)], and transient broadening (29). From the exciton PB in the PM spectrum and strong PL emission from PEPI (Fig. 1D), we conclude that the primary photoexcitations in this 2D perovskite are excitons, mainly 1s excitons. We can therefore study the electron bands simply from the exciton transition into the adjacent continuum band(s), as schematically depicted in Fig. 3C.

The PA1 band from the excitons into the continuum band peaks at \( (350 \pm 2) \text{ meV} \) (the uncertainty comes from the 150-fs pulse duration), which cannot be ascribed to the vertical transition from the 1s exciton into the lowest continuum band because from our EA studies, a transition into the lowest continuum band should appear at 190 meV. We therefore assign PA1 to an optical transition from the 1s exciton to a
second, upper electron continuum branch, which is split from the lower band by Rashba SOC, as shown schematically in Fig. 3C. We have looked for a second PA band at ≈0.2 eV (that is, into the lower continuum band) but could not find any hint of it down to 0.23 eV (Fig. 3A, inset). Our finding suggests that a direct optical transition from the exciton into the lower Rashba split branch is either forbidden or much weaker compared to that in traditional semiconductors such as GaAs (30). To rationalize this result, we note that the transition from a 1s exciton to a band minimum of free electron–hole pairs is proportional to the electrical dipole moment matrix element between a CB (or VB) state and itself. This is because the initial exciton includes a Bloch state of the CB minimum, and the final electron–hole pair includes this same Bloch state. This matrix element is nothing more than the momentum expectation value of the Bloch state, which vanishes at the band minimum. In previous analysis of the exciton-CB transition in GaAs (30), the authors replaced the free carrier wave function by a p-state of an unbound electron–hole pair wave function, enabling the dipole transition exciton → continuum band. In our 2D perovskite film, disorder from point defects and grain boundaries may destroy the coherence of the unbound electron–hole pair, so that optical transitions rely solely on the properties of the Bloch wave functions. Figure 4D shows the density functional theory (DFT) calculation of various electrical dipole transition matrix elements, which verifies that only transitions into the upper Rashba split CB are allowed (see the Supplementary Materials).

We can now obtain the Rashba splitting energy, $E_R$, from the PA$_1$ band at 350 meV, because PA$_1$ should be pushed to higher energies by an energy, $\Delta E = 4E_R$, namely, $E(\text{PA}_1) = E_R + 4E_R$ (Fig. 3C). Using this relation, $E_R (\approx 190$ meV from the EA spectrum) and $E(\text{PA}_1) = 350$ meV (from the transient PA spectrum), we determine $E_R = (40 \pm 5)$ meV in 2D PEPI. We note that the PA$_1$ spectrum is asymmetric; this may be due to the transition from the exciton discrete level to the continuum band, where the exciton wave function is spread in $k$-space by a “$k$-localization length,” $\Delta k$, as determined by its localization length $l$ in real space and $\Delta k \approx 1/l$ (see discussion in the Supplementary Materials) (18). In the inset of Fig. 3A, the solid line through the PA$_1$ data points is a fit using the optical transition model for a 2D semiconductor described in eq. S4 (see the Supplementary Materials), from which we obtain the exciton localization length $l \approx 10$ nm (see the Supplementary Materials). This relatively large $l$ value indicates that the exciton is quite delocalized in the quantum well in a direction perpendicular to the barriers but localized in $k$-space. In addition, the weak PA$_2$ band at $\approx 1$ eV may be readily explained as the optical transition from the exciton level at $\approx 2.4$ eV into higher-lying bands at $\approx 3.3$ eV, which can be seen in the absorption spectrum (fig. S2), similar to the transient PM spectra of excitons in nanotubes and in CH$_3$NH$_3$Pbl$_2$ (31).

Furthermore, we expect that the presence of Rashba splitting would affect the process that leads to free carrier absorption (FCA). To investigate this assumption, we studied the properties of long-lived photoexcitations in PEPI using the technique of steady-state PM (see Materials and Methods). The long-lived photoexcitations should be free carriers because the excitons have a sufficiently long time to ionize into free electrons and holes that can contribute to photocarriers, especially at grain boundaries. This has been verified in PV cells based on 2D perovskites that have shown power conversion efficiency larger than 10% (16). In this case, the PM spectrum would be due to photogenerated FCA.

To verify that we can measure photoinduced FCA by our PM technique, we measured, as a “control experiment,” the steady-state PM spectrum in crystalline Si at 45 K, as seen in Fig. 5A. We could readily fit this PA spectrum by a Drude-type FCA response, in which the PA spectrum varies as $\omega^2$ (32); this validates our approach. In contrast, the steady-state PA spectrum (PA$_{FCA}$) in PEPI film shows a sharp dip at low photon energy, forming a peak at $h\nu \approx 0.15$ eV. We consider this surprising FCA response as a “smoking-gun” verification of the
Rashba splitting that exists in the PEPI continuum band. Because optical transitions within the same branch can only be Drude-like, which is allowed because of the mixture of $s$-state and $p$-state in the CB (see the Supplementary Materials), we ascribe this PA band to which is allowed because of the mixture of $s$-state and $p$-state in the CB, which, as discussed above, vanish at the minimum of the energy dispersion. The solid green line denotes the IB matrix element, which does not vanish at this point, indicating that the optical transition is caused by SOC. The unit cell vectors $a$ and $b$ lie in the $x$-$y$ plane with an angle of 99.7° between them. (B) Schematic of the CB energy dispersion near the R point in the Brillouin zone, where $k_{(12)}$ is directed along the $a + (-b)$ direction. (C) Electronic band structure near the R point, which shows the Rashba splitting along a direction perpendicular to the symmetry-breaking direction; $c_1$ and $c_2$ represent the lower and upper Rashba bands, respectively. (D) DFT-calculated momentum matrix elements versus $k$ near the band minimum (at $k_0 = 0.07 \text{Å}^{-1}$ away from the R point along the $(1,-1)$ direction. Red and blue lines correspond to $x$ and $y$ component of the momentum matrix element between lowest CB $c_1$ and itself, showing the vanishing transition between the exciton and lowest Rashba split CB at $k = k_0$. The green curve is the $z$ component of the momentum matrix element between the Rashba split bands $c_1$ and $c_2$, which is nonzero for all $k$. The $y$ axis is dimensionless, with the computed momentum $p$ presented in terms of its value in Rydberg units: $p_0 = 1.99 \times 10^{-24} \text{kg/(m·s)}$.

To help understand the origin of the Rashba spin-orbit splitting of the CB, we carry out first-principles DFT calculations using local density approximation (LDA) in the form of ultrasoft pseudopotentials, as implemented using Quantum ESPRESSO (see the Supplementary Materials). Figure 4A shows the geometry of the relaxed structure. We find that the inversion symmetry is broken due to the Pb atom displacement from the octahedral center. The displacement is in the 2D plane, roughly in the direction of $a + b$, where $a$ and $b$ are the in-plane lattice vectors. This leads to the Rashba band splitting for states with crystal momentum oriented perpendicularly to the symmetry-breaking direction, as shown in Fig. 4C. The dashed red and blue lines in Fig. 4D show the optical transition matrix elements within the lowest CB, which, as discussed above, vanish at the minimum of the energy dispersion. The solid green line denotes the IB matrix element, which does not vanish at this point, indicating that the optical transition is allowed. An effective tight-binding model given in the Supplementary Materials shows that the source of this optical transition is sp-hybridization present in the CB eigenstates at the band minimum. From our model calculation, we obtain band splitting energy, $E_b = 160 \text{meV}$, which is larger than the experimentally determined splitting. We attribute this discrepancy to approximations used in LDA. For more quantitatively accurate description of the electronic structure, a calculation at the level of quasi-particle GW is likely required.
The 2D hybrid perovskite films were fabricated in a nitrogen-filled glovebox with oxygen and moisture levels of <1 part per million. We mixed R-NH$_2$I (where R = C$_6$H$_5$C$_2$H$_4$I) and PbI$_2$ in a 2:1 molar ratio in N,N’-dimethylformamide to form solutions with a concentration of 0.5 mol/ml. The solutions were spin-coated on an oxygen plasma-pretreated sapphire or KBr substrates at 420 rad/s and 90 s to form 100-nm thick films; the obtained films were subsequently annealed at 100°C for 30 min. The purchased chemicals were used without further purification. From the high transmission through the film and the observation of numerous diffraction orders (fig. S2), we concluded that our spectroscopies were performed on high-quality films.

Transient PM spectroscopy

The main technique in the present study for measurement of the primary photexcitation in the 2D hybrid perovskites was transient PM spectroscopy using the pump-probe correlation method. In this technique, the pump pulse excites the sample, and the mechanically delayed probe pulse measures the pump-induced changes, $\Delta T(t)$, in the sample transmission $T$. The PM spectrum contains PA bands with $\Delta T < 0$ due to excited-state absorption and PB with $\Delta T > 0$ caused by pump-induced bleaching of the ground-state absorption. We used two laser systems based on a Ti:Sapphire oscillator: a low-power (energy per pulse $\approx 0.1$ nJ) high repetition rate ($\approx 80$ MHz) laser for the mid-IR spectral range and a high-power (energy per pulse $\approx 10$ mJ) low repetition rate ($\approx 1$ kHz) laser for the near-IR/visible spectral range. The pump excitation for both laser systems was set at $\hbar \omega = 3.1$ eV. For the low-intensity measurements in the mid-IR spectral range, we used an optical parametric oscillator (Opal, Spectra-Physics) that generates $\hbar \omega$ (probe) from 0.25 to 1.05 eV; for the high-intensity measurements, white-light supercontinuum was generated for $\hbar \omega$ (probe) ranging from 1.15 to 2.7 eV. The transient PM spectra from the two laser systems were normalized to each other using the fundamental (1.55 eV) probe from the low-power laser system. For the PM spectrum in the visible spectral range, we used a Si photodiode, whereas for the mid-IR spectral range, we used an InSb photodiode that was cooled to 80 K to minimize the dark electrical noise.

EA measurements

We used a 2D perovskite film fabricated on a substrate with patterned metallic electrodes. The EA substrate consisted of two interdigitated sets of a few hundreds of 10-μm-wide gold electrodes, which were patterned on a sapphire substrate. The device was placed in a cryostat for low-temperature measurements. By applying a potential, $V$, to the electrodes, a typical electric field, $F \approx 10^8$ V/cm, was generated, with $V = 300$ V and $f = 1$ kHz parallel to the film. To probe the EA spectrum, we used an incandescent light source from a Xe lamp, which was dispersed through a monochromator, focused on the sample, and detected by an ultraviolet-enhanced silicon photodiode. We measured the changes, $\Delta T$, in the transmission spectrum $T$ using a lock-in amplifier, set to twice the frequency ($2f$) of the applied field, and verified that no EA signal was observed at $f$ or $3f$. $\Delta T$ and $T$ spectra were measured separately using a Si photodiode, and the EA spectrum was obtained from the ratio $\Delta T/T$.

Continuous-wave PM measurements

The excitation pump was provided by a diode laser with $h\omega = 2.8$ eV, and the probe beam was provided by an incandescent tungsten/halogen lamp (for the visible/near-IR) or a globar light source (for mid-IR). The sample films grown on KBr substrates were put in a cryostat under vacuum. The sample temperature was varied from 50 to 300 K. The pump and probe beams were overlapped on the sample films, and the transmitted probe beam was filtered through a monochromator and detected by a Si, InGaAs, or HgCdTe detector for different probe spectral ranges. The transmission $T$ and the change of the transmitted probe beam ($\Delta T$), which was caused by the modulated pump beam, were detected by a lock-in amplifier. The PA spectrum was then calculated from $\Delta T/T$.
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