Stable and sensitive amino-functionalized graphene/polyaniline nanofiber composites for room-temperature carbon dioxide sensing

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This article describes the preparation and characterization of amino-functionalized graphene (AmG)/polyaniline (PANI)/poly(methyl methacrylate) (PMMA) nanofiber mats along with the efficiency of these nanofiber composites as a new material for sensing carbon dioxide (CO₂) gas. The surfaces of the PMMA nanofibers were treated at room temperature by ultraviolet (UV) radiation. AmG/PANI was then deposited on the surfaces of the PMMA nanofibers via chemical oxidizing polymerization. It was concluded that UV radiation reduced the hydrophobicity of the PMMA surface through introducing oxidized groups onto the surface. The electrical response of the gas sensor based on the composite nanofibers was investigated at room temperature using various concentrations of CO₂ gas. Compared to the PANI/PMMA nanofibers, the AmG/PANI nanofiber composites displayed a better electrical resistance response to CO₂ at room temperature; the AmG/PANI nanofiber composites exhibited higher sensitivity and faster response times under the same conditions.

Introduction

Carbon dioxide (CO₂) is used in numerous areas such as analytical chemistry, environmental processes, medical diagnostics, and industrial processing. Air pollution is a worldwide concern, and the continuous increase of CO₂ in the atmosphere contributes to global warming, resulting in melting glaciers and the associated rise in sea levels. Moreover, human exposure to high concentrations of CO₂ in the ambient environment causes suffocation and unconsciousness. Thus, researchers are interested in monitoring CO₂ levels in both the ambient (outdoor) and household (indoor) environments. Gas sensors for CO₂ detection are typically based on the change in electrical responses upon exposure to CO₂. As such, research to develop sensing materials has focused on manufacturing high-performance sensing materials and increasing the efficiencies of gas-sensing elements using materials capable of detecting abnormal concentrations of CO₂ in the atmosphere. Therefore, a variety of materials have been considered for application in gas sensors, including inorganic semiconductors, metal oxides, dyes, conducting polymers, and carbon nanomaterials. Among these sensing materials, graphene, a two-dimensional monolayer of carbon atoms, has been identified as a promising sensing material owing to its exceptional chemical and electronic properties, mechanical stiffness, and electrical conductivity, which are desirable properties for fabricating resistive-type CO₂ gas sensors. Research on graphene has focused on understanding the extensive range of electronic, optical, thermal, and mechanical properties of graphene materials. Nevertheless, numerous features of graphene can only be exploited when the graphene is integrated into more complex assemblies. The chemical functionalization of graphene is the most common technique used to create these assemblies since it enables chemical bonding between graphene and the material of interest (e.g., small molecules or polymer chains). Moreover, the chemical functionalization of graphene is a particularly attractive because it can improve the solubility and processability of graphene.

Graphene can be functionalized with organic, monomeric amines consisting of primary, secondary, or tertiary amine groups, which are known to be sensitive to CO₂ gas. The binding of CO₂ to amino groups is based on the rule of hard and soft acids and bases. CO₂ is a hard acid and can efficiently interact with the amino groups, which are hard bases, to form carbamates or bicarbonates, resulting in an increase in resistance. In this work, we used ethylenediamine [NH₂(CH₂)₂NH₂] to functionalize the graphene surfaces, similar to in our previous work. Many studies have demonstrated that primary amine groups are highly efficient in adsorbing CO₂ gas. Therefore, in this work, the fundamental mechanism of CO₂ sensing is the efficient binding of amino-functionalized graphene with CO₂ at room temperature (RT) to form carbamates via a reversible reaction.
Polyaniline (PANI) has been widely studied for gas detection based on its high sensitivity, fast response time, and functionality at room temperature. However, the poor cycling stability of PANI-based sensors remains a significant obstacle for practical applications. However, the incorporation of conductive carbon materials can improve the cycle lifespan of PANI-based sensors. Combining the excellent sensing properties of amine-functionalized graphene (AmG) and PANI in an AmG/PANI composite is expected to enhance the stability, sensitivity, and selectivity of the resulting sensor. Poly(methyl methacrylate) (PMMA) is used in a wide range of nanotechnology applications. PMMA presents several advantages over other polymers in this study, including high mechanical strength, good thermal stability, nontoxicity, solubility in a variety of solvents, and good dielectric properties. These characteristics result in a polymer solution with good conductivity, which is favorable for electrospinning. Therefore, the sensitivity and recovery of the sensor could be improved via the in situ polymerization of AmG/PANI onto the surfaces of flexible electrospun PMMA substrates. The resulting nanofiber mats have an extremely high surface area-to-volume ratio and porosity, both of which are desirable properties in sensor applications.

Based on our literature survey in the field of composite-based nanofiber CO\textsubscript{2} sensors, no work has been reported on AmG/PANI nanofiber composite-based CO\textsubscript{2} sensors. In this study, PMMA nanofiber mats were produced via electrospinning and then treated with ultraviolet (UV) radiation at a wavelength of 365 nm. Subsequently, AmG/PANI were in situ polymerized on the surfaces of electrospun PMMA nanofibers to obtain flexible, composite nanofibers for CO\textsubscript{2} detection. The resulting AmG/PANI/PMMA sensor showed high sensitivity toward CO\textsubscript{2} gas at a concentration of 20 ppm. In addition, the response was reproducible for CO\textsubscript{2} gas concentrations ranging from 20 to 2000 ppm.

**Results and discussion**

**Effect of UV treatment on substrate morphology**

The SEM image in Fig. 1a shows the morphology of the neat PMMA nanofibers obtained from electrospinning. Nanofibers with smooth surfaces are randomly distributed in the membrane with relatively uniform sizes of 685 ± 220 nm. After the deposition of AmG/PANI by in situ polymerization (Fig. 1b), polymerization occurred only on the PMMA nanofiber surfaces and did not infiltrate the PMMA nanofibers due to the hydrophobicity of the PMMA nanofiber surfaces. Therefore, the PMMA nanofibers were treated by UV radiation (wavelength = 365 nm) for 5 min (each side). UV radiation modified the

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![Fig. 1](image-url) (a) An SEM image of PMMA nanofibers and the distribution of their diameters (the inset shows an SEM image at higher magnification). (b) An SEM image of PMMA after the deposition of AmG/PANI. (c) An SEM image of PMMA after UV treatment and the distribution of PMMA fiber diameters (the inset shows an SEM image with higher magnification). (d) An SEM image of PMMA nanofibers after UV treatment and the deposition of AmG/PANI.
surface by oxidizing it, resulting in carboxylic acid groups, and increased the radical oxygen content of the exposed surface, causing the nanofiber surface to become more hydrophilic.\textsuperscript{30–32} While UV treatment modified the PMMA nanofiber surfaces, no remarkable changes in nanofiber morphology (668 ± 209 nm) were observed (Fig. 1c). As shown in Fig. 1d, AmG/PANI was uniformly dispersed in the PMMA nanofibers, indicating that UV treatment reduced the hydrophobicity of the PMMA surface by introducing oxidized groups onto the nanofiber surfaces. Similar results regarding the surface oxidation of PMMA have been previously reported.\textsuperscript{33–35}

Contact angle measurements were performed on the PMMA nanofibers before and after UV treatment to study the surface wettability of the PMMA nanofibers. As shown in Fig. 2, the contact angle of PMMA nanofibers before UV irradiation was 133.1°. After UV treatment, the contact angle was significantly decreased to 85.3°, indicating better wettability. The results suggest that UV radiation enhanced the wettability of the PMMA nanofibers by introducing active sites onto the nanofiber surfaces.\textsuperscript{29} These results agree with the SEM results discussed above.

Gas sensing properties of AmG/PANI nanofiber composites

A flexible nanofiber composite sensor was placed in the sensing chamber to observe its sensing behavior toward CO\textsubscript{2} gas at room temperature. Fig. 3 shows the response of the AmG/PANI nanofiber composite. The response increased dramatically when exposed to various concentrations of CO\textsubscript{2} ranging from 20 to 2000 ppm and recovered to the original value when the CO\textsubscript{2} was replaced by N\textsubscript{2}.

The sensing results are similar to other results reported for CO\textsubscript{2} sensors based on π-conjugated amine (NBA) and zinc oxide (ZnO) nanohybrids,\textsuperscript{25} polyethylenimine,\textsuperscript{27} graphene (G) and aluminum oxide (Al\textsubscript{2}O\textsubscript{3}) quantum dot composites,\textsuperscript{38} G/antimony oxide (Sb\textsubscript{2}O\textsubscript{3}) quantum dot composites,\textsuperscript{39} and G/PANI/polystyrene (PS) nanocomposite films.\textsuperscript{40} The high sensitivity of the AmG/PANI nanofiber composite for CO\textsubscript{2} is attributed to the formation of the carbamate moiety, which is the basis of the reversible CO\textsubscript{2} response that occurs more efficiently at lower temperatures.
toward 20 ppm of CO\textsubscript{2} at room temperature. The PANI nanoﬁber-based sensor exhibited no sensitivity to CO\textsubscript{2} gas at 20 ppm, unlike the sensor based on AmG/PANI nanoﬁbers. Fig. 4b shows the response and recovery characteristics of a single cycle for the AmG/PANI nanoﬁber gas sensor for 20 ppm of CO\textsubscript{2} gas. The time required for resistance to increase to 90\% of the highest value is known as the response time of the gas sensor. The time required for the resistance to decrease from 90\% of the highest value to the initial resistance value is known as the recovery time. The response and recovery times of the AmG/PANI nanoﬁber gas sensor were approximately 10 and 20 s, respectively, much better than those reported for other CO\textsubscript{2} sensors (Table 1).

Table 1 shows that our sensor showed a higher sensitivity toward CO\textsubscript{2} at room temperature compared to other reported CO\textsubscript{2} sensors, with the exception of G/PANI/PS nanocomposite sensors. However, our response and recovery times were much better, as mentioned above.

The bar plot in Fig. 5 illustrates the selectivity of the sensor for the detection of CO\textsubscript{2} over carbon monoxide (CO), ammonia (NH\textsubscript{3}), and hydrogen (H\textsubscript{2}) at concentrations of 100 ppm. The sensor showed extreme selectivity to CO\textsubscript{2} compared to the other gases at the same concentration. Mandal et al.\textsuperscript{25} reported a high selectivity at 5000 ppm using NH\textsubscript{3}, CO, and H\textsubscript{2}S gases. A similar result was obtained in this study but at a considerably lower ppm.

The stability of the AmG/PANI nanoﬁber gas sensor was examined for 30 days. Good stability is essential for the use of the sensor in practical applications. Fig. 6 presents the stability of the AmG/PANI sensor exposed to 300 ppm of CO\textsubscript{2} for 30 days. The stability was evaluated each day for 10 days. After observing no change in sensitivity for 10 days, the stability was then tested every three days. The results clearly indicate that the sensor exhibited nearly constant sensing signals for one month.

The excellent sensing properties of the AmG/PANI nanoﬁber composite toward CO\textsubscript{2} gas can be attributed to the following two mechanisms. Usually, of the three classes of amine, primary amines react with CO\textsubscript{2} to produce carbamates. The reaction mechanism of CO\textsubscript{2} with primary amine to form carbamates is shown in eqn (1):

$$\text{CO}_2 + 2\text{RNH}_2 \leftrightarrow \text{RNHCOO}^- + \text{RNH}_3^+.$$  \hspace{1cm} (1)

Theoretically, two primary amines are needed to trap CO\textsubscript{2}.\textsuperscript{24} First, a lone pair of amine electrons attaches to the carbon atom of CO\textsubscript{2} to form a zwitterion. Next, another free amine deprotonates the zwitterion to form the carbamate. Caplow et al.\textsuperscript{36} were the first to describe the zwitterionic mechanism in the formation of carbamate from the interaction between the primary amine and CO\textsubscript{2}. During this reaction, the number of free amines is reduced, and the proton mobility is subsequently reduced, which in turn increases the resistance.\textsuperscript{37,38} In the second mechanism, the p-type PANI and n-type AmG in the AmG/PANI nanoﬁber composite form a p–n junction at the interface between them.\textsuperscript{39,40} Consequently, a depletion layer is

**Table 1** Comparison of the performances of the AmG/PANI sensor with other sensors reported in the literature

| Material                   | CO\textsubscript{2} concentration (ppm) | Response time (s) | Recovery time (s) | Temp. (\^\circ\textsubscript{C}) | Ref.   |
|---------------------------|----------------------------------------|-------------------|-------------------|----------------------------------|-------|
| AmG/PANI nanoﬁber composite | 20                                     | 10                | 20                | RT                               | This work |
| NBA/ZnO nanohybrids       | 500                                    | 206               | 354               | RT                               | 25    |
| G/Sb\textsubscript{2}O\textsubscript{3} composite | 50                                     | 16                | 22                | RT                               | 42    |
| G/PANI/PS nanocomposite    | 20                                     | 65                | 65                | RT                               | 43    |
| Carbon nanotube (CNT) thin film | 800                              | >20               | >75               | RT                               | 44    |

Fig. 5 The selectivity of the AmG/PANI nanoﬁber composite gas sensor to various gases at concentrations of 100 ppm.

Fig. 6 The long-term stability of the AmG/PANI nanoﬁber composite gas sensor when exposed to 300 ppm CO\textsubscript{2}.
formed at the interface.\textsuperscript{40,41} Fig. 7 schematically shows the formation of the p–n junction between p-type PANI and n-type AmG. When CO$_2$ gas is injected into the chamber, the gas molecules are adsorbed by the surface of the AmG/PANI nanofiber composite. When the electrons are released at the junction into PANI, the majority carriers (holes) in PANI are compensated, which might increase the thickness of the depletion layer. This further increases the resistance of the AmG/PANI sensor in the presence of CO$_2$.\textsuperscript{25,41} Thus, the enhanced CO$_2$ sensing properties of the AmG/PANI nanofiber sensor can be attributed to the presence of the primary amine functional group and the p–n junction between p-type PANI and n-type AmG.

Conclusions

In summary, we designed and synthesized a CO$_2$ gas sensor via chemical polymerization using PMMA nanofibers as a flexible substrate and AmG/PANI as an active material. The fabricated sensors exhibited good CO$_2$ sensing properties at room temperature, fast response and recovery times (10 and 20 s, respectively), and good stability. The easy preparation and excellent properties of AmG/PANI suggest the viability of electrospun nanofibers as gas sensing materials. The sensing mechanisms of the AmG/PANI nanofiber sensor toward CO$_2$ were discussed. The results demonstrate that AmG/PANI/PMMA nanofiber composites are promising candidates for practical CO$_2$ sensors.

Experimental section

Materials

AmG was synthesized in our lab. GO, aniline (ACS reagent, \textgeq 99.5%), N,N-dimethylformamide (DMF, 99.8%), PMMA (M$_w$ = 996 000 g mol$^{-1}$), ethylenediamine (EDA, \textgeq 99%), ammonium persulfate (APS, \textgeq 98.0%), and 5-sulfosalicylic acid dihydrate (SSA, \textgeq 99%) were obtained from Sigma-Aldrich (Oakville, ON, Canada). Deionized (DI) water was used in all experiments.

Synthesis of amino-functionalized graphene

GO was reduced by thermal annealing, as reported previously.\textsuperscript{27} Briefly, reduced GO (200 mg) was dispersed in 15 mL of EDA in a vessel under vigorous stirring for 24 h under reflux at 80 °C. The AmG was then isolated by centrifugation and washed thoroughly with DI water, filtered, and dried in a vacuum oven at 80 °C for 24 h.

Preparation of PMMA nanofibers by electrospinning

A homogeneous dispersion of 1.06 g PMMA in 20 g DMF was electrospun using a homemade horizontal setup (see Fig. 8a). The setup consisted of a 5 mL glass syringe, a metallic needle (inner diameter = 0.84 mm; Cadence Science, Cranston, RI, USA), a syringe pump (PHD 2000; Harvard Apparatus, Holliston, MA, USA), and a high-voltage power supply (ES60P-5W; Gamma High Voltage Research Inc, Omaha Beach, FL, USA). The PMMA solutions were electrospun at room temperature for 24 h using a positive voltage of 27 kV, a working distance of 20 cm (distance between the needle tip and the collecting plate), and a solution feed rate of 0.6 mL h$^{-1}$. The nanofibers were collected continuously until mats of nanofibers with a thickness of approximately 0.75 mm were formed.

UV radiation treatment.

The PMMA electrospun nanofibers were subjected to UV radiation at a wavelength of 365 nm for different exposure times (10 and 20 min). The radiation time of 10 min was chosen for subsequent experiments.

Fig. 7 A schematic illustration of the formation of the p–n junction between p-type PANI and n-type AmG.

Fig. 8 (a) A schematic illustration of the electrospinning setup consisting of a power supply, syringe, and collecting collector. (b) Homemade setup for UV irradiation.

Fig. 9 A schematic illustration of the fabrication process of the AmG/PANI/PMMA nanofiber sensor.
Sensor fabrication of sensors

Treated and untreated PMMA nanofiber mats were fixed to Teflon films (25 × 25 mm) using adhesive tape. AmG/PANI was then grown on the surface of electrospun PMMA nanofibers via in situ polymerization. Fig. 9 schematically illustrates the fabrication process of the AmG/PANI/PMMA nanofiber composite sensors. First, 10 mg AmG was dispersed in 50 mL DI water by sonication for 1 h. Next, 2.45 g SSA and 1.86 g aniline were dissolved and mechanically stirred for 30 min to form solution A, while 4.45 g APS was dissolved in 50 mL DI water and stirred constantly for 30 min to form solution B. Solutions A and B were kept for 1 h in the refrigerator at 5 °C. Subsequently, the PMMA nanofiber mats were immersed in solution A. Solution B was then added dropwise into solution A followed by stirring in an ice-water bath for different polymerization times (1, 2, and 4 h). The polymerization time of 1 h was chosen for further experiments. Subsequently, the nanofiber mats were removed from the solution, washed with a large amount of DI water, and washed several times with ethanol to remove the residual oxidant. Finally, all nanofiber composites were dried at room temperature. Solutions A and B solutions were similarly prepared without the addition of AmG for comparative analysis. Copper tape was used as an electrode on the two opposite ends of a Teflon film for electrical connection. The sensors were stored at room temperature under vacuum until their sensing properties were tested.

Measurements

To study the morphologies of the PMMA nanofibers before and after UV treatment, the PANI/PMMA and AmG/PANI/PMMA nanofiber composites were examined using a table-top scanning electron microscope (TM3030 plus, Toshiba, Rexdale, ON, Canada) without any metallic coating. The fiber diameters were determined using Image-Pro Plus® software. Approximately 900 nanofibers were randomly selected from three independent SEM images (300 from each specimen) and used for analysis. Contact angle measurements were carried out using the sessile drop method at room temperature under atmospheric conditions with n OCA contact angle system (DataPhysics Instruments GmbH, Filderstadt, Germany) in combination with SCA 20 software (Dataphysics Instruments) for image analysis and contact angle calculation. A small volume (2 μL) of water was pipetted onto the nanofiber surface. Images were automatically taken as function of time. The reported contact angles are the average of five measurements taken at different positions on each sample.

To characterize the performance of the fabricated AmG/PANI/PMMA nanofiber composites as a CO2 sensor, the sensor resistance values were recorded using a PalmSens3 instrument (PalmSens EmStat+Potentiostat w/Bluetooth, Compact Electrochemical interfaces, BASi®, West Lafayette, IN, USA) upon exposure to CO2 gas at different concentrations (see Fig. 10). The flow rates of CO2 and nitrogen were controlled by mass flow controllers (MKS instruments Inc., 1179C mass-flow®, Kanata, ON, Canada). The concentration of CO2 gas was varied from 20 to 2000 ppm by adjusting the flow rates of CO2 and nitrogen. The sensitivity of the sensor was investigated in a square, sealed, glass test chamber with a gas inlet and gas outlet along with electrical connections. When the sensor resistance was stable, CO2 gas was injected into the test chamber via a micro-injector through a rubber plug. After the resistance reached a new value, nitrogen gas was injected into the test chamber for gas sensor recovery. The sensitivity (response) of the sensor was defined as the ratio of the change in the resistance $S$:

$$S(\%) = \left( \frac{R_{\text{gas}} - R_{\text{N}_2 \text{or air}}}{R_{\text{N}_2 \text{or air}}} \right) \times 100,$$  

where $R_{\text{N}_2 \text{or air}}$ is the initial resistance obtained using N2 or air as the base gas, respectively, and $R_{\text{gas}}$ is the resistance of the sensor in CO2.

The response (recovery) time is the time taken for the sensor to achieve a minimum of 90% of the total resistance change in the case of adsorption (desorption). All experiments were conducted at room temperature, and the relative humidity within the chamber was the same as the ambient relative humidity, which was between 30% and 35% during measurement.

Author contributions

Hanan Abdali designed the study, performed the experiments, collected test data, interpreted results, and drafted the manuscript. Bentolhoda Heli helped perform the experiments and interpret the results. Abdellah Ajji designed the study, interpreted the results, and reviewed and edited the manuscript.

Conflicts of interest

There are no conflicts to declare.

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References

1. C. E. Booth, D. G. McDonald and P. J. Walsh, Acid-base balance in the sea mussel, Mytilus edulis. 1. Effects of hypoxia and air-exposure on hemolymph acid-base status, *Mar. Biol. Lett.*, 1984, 5, 347–358.

2. R. Desai, et al., Indium sesquitelluride (In$_2$Te$_3$) thin film gas sensor for detection of carbon dioxide, *Sens. Actuators, B*, 2005, 107(2), 523–527.

3. G. G. Mandayo, et al., Performance of a CO$_2$ impedimetric sensor prototype for air quality monitoring, *Sensors*, 2011, 11(5), 5047–5057.

4. J. Herrán, G. G. Mandayo and E. Castano, Solid state gas sensor for fast carbon dioxide detection, *Sens. Actuators, B*, 2008, 129(2), 705–709.

5. C.-J. Chiang, et al., *In situ* Fabrication of conducting polymer composite film as a chemical resistive CO$_2$ gas sensor, *Microelectron. Eng.*, 2013, 111, 409–415.

6. C. Wang, et al., Metal oxide gas sensors: sensitivity and influencing factors, *Sensors*, 2010, 10(3), 2088–2106.

7. S. Park, C. Park and H. Yoon, Chemo-electrical gas sensors based on conducting polymer hybrids, *Polymers*, 2017, 9(5), 155.

8. E. Llobet, Gas sensors using carbon nanomaterials: a review, *Sens. Actuators, B*, 2013, 179, 32–45.

9. T. T. Tung, et al., Recent advances in sensing applications of graphene assemblies and their composites, *Adv. Funct. Mater.*, 2017, 27(46), 1702891.

10. K. S. Novoselov, et al., Two-dimensional gas of massless Dirac fermions in graphene, *Nature*, 2005, 438(7065), 197.

11. K. S. Novoselov, et al., Electric field effect in atomically thin carbon films, *Science*, 2004, 306(5696), 666–669.

12. C. Lee, et al., Measurement of the elastic properties and intrinsic strength of monolayer graphene, *science*, 2008, 321(5887), 385–388.

13. A. A. Balandin, et al., Superior thermal conductivity of single-layer graphene, *Nano Lett.*, 2008, 8(3), 902–907.

14. J. Du and H. M. Cheng, The fabrication, properties, and uses of graphene/polymer composites, *Macromol. Chem. Phys.*, 2012, 213(10–11), 1060–1077.

15. T. Kuila, et al., Chemical functionalization of graphene and its applications, *Prog. Mater. Sci.*, 2012, 57(7), 1061–1105.

16. D. Cai and M. Song, Recent advance in functionalized graphene/polymer nanocomposites, *J. Mater. Chem.*, 2010, 20(37), 7906–7915.

17. C. Bao, et al., *In situ* preparation of functionalized graphene oxide/epoxy nanocomposites with effective reinforcements, *J. Mater. Chem.*, 2011, 21(35), 13290–13298.

18. M. Ionitiată, et al., Graphene and functionalized graphene: extraordinary prospects for nanobiocomposite materials, *Composites, Part B*, 2017, 121, 34–57.

19. J. Xu, Y. Wang and S. Hu, Nanocomposites of graphene and graphene oxides: synthesis, molecular functionalization and application in electrochemical sensors and biosensors. A review, *Microchim. Acta*, 2017, 184(1), 1–44.

20. K. Hu, et al., Graphene-polymer nanocomposites for structural and functional applications, *Prog. Polym. Sci.*, 2014, 39(11), 1934–1972.

21. H. J. Salavagione, Promising alternative routes for graphene production and functionalization, *J. Mater. Chem. A*, 2014, 2(20), 7138–7146.

22. V. Georgakilas, et al., Functionalization of graphene: covalent and non-covalent approaches, derivatives and applications, *Chem. Rev.*, 2012, 112(11), 6156–6214.

23. Y. G. Ko, S. S. Shin and U. S. Choi, Primary, secondary, and tertiary amines for CO$_2$ capture: designing for mesoporous CO$_2$ adsorbents, *J. Colloid Interface Sci.*, 2011, 361(2), 594–602.

24. S. Stegmeier, et al., Mechanism of the interaction of CO$_2$ and humidity with primary amino group systems for room temperature CO$_2$ sensors, *Procedia Chem.*, 2009, 1(1), 236–239.

25. B. Mandal, et al., π-Conjugated Amine–ZnO Nanohybrids for the Selective Detection of CO$_2$ Gas at Room Temperature, *ACS Appl. Nano Mater.*, 2018, 1(12), 6912–6921.

26. S.-A. Chen and G.-W. Hwang, Structures and properties of the water-soluble self-acid-doped conducting polymer blends: sulfonic acid ring-substituted polyaniline/poly(vinyl alcohol) and poly (aniline-co-N-propanesulfonic acid aniline)/poly(vinyl alcohol), *Polymer*, 1997, 38(13), 3333–3346.

27. H. Abdali and A. Aji, Preparation of Electrospun Nanocomposite Nanofibers of PolyvinylidePoly(methyl methacrylate) with Amino-Functionalized Graphene, *Polymers*, 2017, 9(9), 453.

28. A. Polini, et al., Collagen-functionalised electrospun polymer fibers for bioengineering applications, *Soft Matter*, 2010, 6(8), 1668–1674.

29. A. Rabiatul, et al., Surface modification of electrospun poly(methyl methacrylate) (PMMA) nanofibers for the development of in vitro respiratory epithelium model, *J. Biomater. Sci., Polym. Ed.*, 2015, 26(17), 1297–1311.

30. I. Pashkuleva, et al., Surface modification of starch based biomaterials by oxygen plasma or UV-irradiation, *J. Mater. Sci.: Mater. Med.*, 2010, 21(1), 21–32.

31. A. Yusilawati, et al., Surface modification of polystyrene beads by ultraviolet/ozone treatment and its effect on gelatin coating, *Am. J. Appl. Sci.*, 2010, 7(6), 724.

32. J. M. Goddard and J. Hotchkiss, Polymer surface modification for the attachment of bioactive compounds, *Prog. Polym. Sci.*, 2007, 32(7), 698–725.

33. Y. Dekhtyar, et al., Surface properties of ocular prostheses material change under UV influence, in *5th European Conference of the International Federation for Medical and Biological Engineering*, Springer, 2011.

34. A. Nematollahzadeh, et al., Increasing the interfacial adhesion in poly(methyl methacrylate)/carbon fibre composites by laser surface treatment, *Polym. Polym. Compos.*, 2006, 14(6), 585–589.
35. D. Rymuszka, et al., Changes in surface properties of polymethylmethacrylate (PMMA) treated with air plasma, *Ann. UMCS, Chem.*, 2015, 70(1), 65–78.
36. M. Caplow, Kinetics of carbamate formation and breakdown, *J. Am. Chem. Soc.*, 1968, 90(24), 6795–6803.
37. S. Choi, J. H. Drese and C. W. Jones, Adsorbent materials for carbon dioxide capture from large anthropogenic point sources, *ChemSusChem*, 2009, 2(9), 796–854.
38. T. C. Doan, et al., Carbon dioxide detection with polyethyleneimine blended with polyelectrolytes, *Sens. Actuators, B*, 2014, 201, 452–459.
39. D. Majumdar, Functionalized-Graphene/Polyaniline Nanocomposites as Proficient Energy Storage Material: An Overview, *Innov. Energy Res.*, 2016, 5(145), DOI: 10.4172/2576-1463.1000145.
40. H. Tai, et al., Fabrication and gas sensitivity of polyaniline–titanium dioxide nanocomposite thin film, *Sens. Actuators, B*, 2007, 125(2), 644–650.
41. J. Sun, et al., Preparation of polypyrrole@WO3 hybrids with pn heterojunction and sensing performance to triethylamine at room temperature, *Sens. Actuators, B*, 2017, 238, 510–517.
42. K. Nemade and S. Waghuley, Role of defects concentration on optical and carbon dioxide gas sensing properties of Sb2O3/graphene composites, *Opt. Mater.*, 2014, 36(3), 712–716.
43. J. Bhadra, et al., Fabrication of polyaniline–graphene/ polystyrene nanocomposites for flexible gas sensors, *RSC Adv.*, 2019, 9(22), 12496–12506.
44. Z.-D. Lin, S.-J. Young and S.-J. Chang, CO2 gas sensors based on carbon nanotube thin films using a simple transfer method on flexible substrate, *IEEE Sens. J.*, 2015, 15(12), 7017–7020.