Research Article

Characterisation of Natural Fibre Reinforcements and Composites

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Received 18 October 2013; Accepted 25 November 2013

Academic Editor: Masamichi Kawai

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Recent EU directives (e.g., ELV and WEEE) have caused some rethinking of the life cycle implications of fibre reinforced polymer matrix composites. Man-made reinforcement fibres have significant ecological implications. One alternative is the use of natural fibres as reinforcements. The principal candidates are bast (plant stem) fibres with flax, hemp, and jute as the current front runners. The work presented here will consider the characterisation of jute fibres and their composites. A novel technique is proposed for the measurement of fibre density. The new rule of mixtures, extended for noncircular cross-section natural fibres, is shown to provide a sensible estimate for the experimentally measured elastic modulus of the composite.

1. Introduction

There have been a number of recent reviews [1–9] of natural fibre reinforcements and their composites. Virk et al. [10] proposed an extension to the rule of mixtures (ROM) for the estimation of the Young’s modulus of a composite with reinforcements of noncircular cross-section:

\[ E_c = \kappa \eta_d \eta_l \eta_o V_f E_f + V_m E_m, \]

(1)

where \( E_c \) is the Young’s modulus, \( V_c \) is the volume fraction, \( \kappa \) is the fibre area correction factor (FACF) [10], \( \eta_d \) is the fibre diameter distribution factor (FDDF) [11], \( \eta_l \) is the fibre length distribution factor (FLDF) [12], and \( \eta_o \) is the fibre orientation distribution factor (FODF) [13] with subscript c, f, m, and v being composite, fibre, isotropic matrix, and voids (\( V_f + V_m + V_v = 1 \)), respectively. For the case of hollow fibres, the voidage should be classified as that internal to the fibre (i.e., the lumen in natural fibres), \( V_v \), and that external to the fibre, \( V_e \), such that \( V_f + V_m + (V_v + V_e) = 1 \). When the fibres are characterised taking into account the internal features, the internal voids are not expected to influence the mechanical properties of the composites. The separation of internal and external voids is equally applicable to man-made hollow fibres [14, 15]. For hollow fibres, it is then necessary to determine the density of the fibre material (primarily cellulose in natural fibres) and the volume fraction of the void within the fibre.

Typical values for each of these parameters would be

(i) \( E_f = 14–87 \) GPa (bast natural fibres) [6], \( \sim 70\) GPa (glass), \( \sim 140 \) GPa (aramid), or \( \sim 210 \) GPa (carbon),

(ii) \( E_m = 1–3 \) GPa (polymers),

(iii) \( V_f = 0.1–0.3 \) (random orientation), 0.3–0.6 (woven fabric), or 0.5–0.8 (unidirectional),

(iv) \( \kappa = 1.42 \) (i.e., \( 2697 \mu m^2/1896 \mu m^2 \)) for jute [10] as the factor compensates for the overestimate in the apparent cross-sectional area (CSA) when CSA is derived from an apparent diameter with the assumption of circular cross-section and is calculated as the ratio of apparent CSA/true CSA,

(v) \( \eta_d = 1 \) for well-characterised fibres but \( \neq 1 \) where there is a dependence of the modulus on fibre diameter,

(vi) \( \eta_l = 0 \) (if significantly less than the critical length as defined by Cox shear-lag theory) to 1 (for continuous fibres),

(vii) \( \eta_o = 1/4 \) (biaxial on the bias angle), 3/8 (random in-plane), 1/2 (biaxial parallel to the fibres), or 1...
(unidirectional parallel to the fibres) as determined by the Krenchel equation:

\[
\eta_\alpha = \sum_{i=0}^{180} V_i \cos^4 \theta_i. \tag{2}
\]

The indicative materials data above should not be used for “design” purposes. Daniel and Ishai [16] have stated that the ROM can be used where the fibre is anisotropic with different properties in the axial and transverse (radial) directions and that the matrix is isotropic. A more complete treatment of this assumption is given by Gibson [17].

The determination of the parameters required for the rule of mixtures can be achieved in a variety of ways including the Grafil [18] and CRAG [19] Test Methods. The determination of the fibre volume fraction in a natural fibre reinforced composite is problematic. The fibres are hygroscopic and so may swell, and weight is a function of the moisture content. The density of the fibre should be quoted as a specific moisture content or against moisture content. The absorption of liquid by the test specimen may complicate the determination of the mass of the fibre. Subject to the above, the fibre volume fraction of composite materials may be obtained directly, or by use of the rule of mixtures for density, \( \rho_f = V_f \rho_f + V_m \rho_m \), by one or more of the following methods:

(i) tow counting for unidirectional composites in an open-ended mould,
(ii) fabric areal weight in a moulding of known thickness,
(iii) direct weighing when no fibre is lost in the moulding process,
(iv) density gradient column,
(v) Archimedes principle using weight measurements in air and in water (or other fluids),
(vi) resin burn-off in an oven at 580–600°C: inappropriate for natural fibre composites,
(vii) thermogravimetric analysis (TGA),
(viii) chemical digestion,
(ix) optical microscopy, electron microscopy or X-ray tomography, and image analysis [20, 21].

More information, and references, for each of these techniques can be found in [7].

Facca et al. [22] found that standard micromechanics models for the prediction of the elastic properties of natural fibre composites had mixed success. This paper will consider a novel route to the accurate determination of the fibre density, and then examine the use of the extended rule of mixtures (1) to predict the tensile modulus of a jute fibre reinforced epoxy composite.

2. Experimental Methodology

Jute fibre of unknown provenance (from the same batch as studied in references [10, 23]) was chosen as the material for study. The fibre density was determined by Archimedes principle. Fibres were coiled into small bundles and dried at 60°C for 30 min and then immediately weighed in air on an Avery-Berker digital scale (serial number 59030902). The fibres were then suspended, immersed in the test fluid, and degassed in a vacuum chamber:

(i) in water (with Ilfotol as a wetting agent) at a vacuum level of −990 mbar,
(ii) in acetone at a vacuum level of −500 mbar (vacuum level limited to prevent sudden boiling).

When all the bundles had sunk to the base of the container and no gas could be seen escaping from the bundle, the chamber was slowly vented to atmosphere and the bundle was weighed on a Stanton Instruments analogue scale (serial number 19635). The fibre density, \( \rho_f \), was calculated using the equation in CRAG method 800:

\[
\rho_f = \frac{a \rho_w}{(a-b)}, \tag{3}
\]

where \( a \) is the mass in air, \( b \) is the apparent mass in the fluid, and \( \rho_w \) is the density of the fluid at the temperature of the test. The densities determined for the jute fibres were

\[
\rho_f = 1669 \pm 37 \text{ kg} \cdot \text{m}^{-3} \text{ in water/Ilfotol at } 22.4°C,
\rho_f = 1652 \pm 37 \text{ kg} \cdot \text{m}^{-3} \text{ in acetone at } 20.3°C. \tag{4}
\]

Note that these densities are higher than those normally quoted in the natural fibre literature (~1525 kg·m⁻³ [6] for flax, hemp, and jute), but the latter are rarely determined from degassed fibres. After removal from the respective fluids, the fibres were allowed to equilibrate with the ambient humidity for thirty minutes and then weighed in air. Absorption of 4% moisture resulted in a density of 1570 ± 43 kg·m⁻³ which is comparable with the highest values quoted in the literature for natural fibres.

The tensile properties of single fibres from jute sliver were obtained using Grafil Test Method 101.13 on an Instron 1026 universal testing machine (serial number H2709) with an Instron 2511-101 500 g load cell (serial number UK953, calibrated with 50 g weights) at a cross-head speed of 0.5 mm/min. The fibre cross-sectional area (CSA) was obtained by embedding fibres from the microscopy area of the test card (Figure 1) along the principal axis of a cylindrical block of epoxy resin. The cast cylinder was split parallel to the

**Figure 1:** Schematic of the Grafil test card (grey) with the fibre (black) and the left aperture for tensile testing (50 mm gauge length) and the right aperture for microscopy (10 mm aperture).
flat surfaces to provide two representative sections separated by 3–4 mm (the width of the slitting saw blade) in respect of their position along the fibre. Digital images of each cross-section were obtained using a Nikon 995 3.2 megapixel digital camera from a Leica DMRB inverted optical transmission microscope and the CSA was determined using ImageJ software (http://rsb.info.nih.gov/ij/) referenced to images of a standard 10000 μm graticule with 10 μm divisions.

To manufacture the composite, 506 g/m (506 ktx) jute spun yarn was wound onto a lathe-mounted spool to produce an aligned quasi-unidirectional reinforcement. This reinforcement was transferred to the mould on a wire frame designed to maintain a slight tension on the fibres. The reinforcement was transferred to the mould on a wire frame designed to maintain a slight tension on the fibres. The composite was manufactured by resin infusion under flexible tooling with a flow medium (RIFT II) [24] allowing two hours under vacuum for the fibre to degas at 9.7 mbar absolute (Digitron 8025P vacuum gauge) before infusion with SP Systems Prime 20 epoxy resin system (target mix ratio = 100 g batch 090730 resin/25 g batch 089017 slow hardener). Posture was 3 h at 25°C, ramp to 60°C at +0.5°C/min, 16 h at 60°C, and ramp at −0.5°C/min to return to ambient temperature.

Mechanical properties of the jute yarn composites were obtained according to ISO 527-5 for tensile modulus and strength on an Instron 1175 universal testing machine (serial number H0718) with an Instron 5 kN load cell (serial number UK3191) at a cross-head speed of 2 mm/min. The FACF (η_f) and FDDF (η_d) were taken to be unity. Fibre lengths were determined by deconstruction of a one metre length of the 506 ktx spun yarn. Over half the fibres by weight had lengths in the range 100–180 mm with <12% by weight being <50 mm and no fibres >280 mm long. It is therefore assumed reasonable to take FLDF (η_f) = 1. The twist angle of the spun yarn was ~30° (ten measurements along 1m of yarn), so the FODF (η_d) = 0.563 using the Krenchel FODF (2). The fibre (V_f) and void (V_v) volume fractions from optical microscopy were 61.2 ± 1.1% and 36 ± 3.3%, respectively. Using (I) with tensile moduli from the single fibre experiments and E_m = 3.2 GPa (SP Systems data), the predicted modulus was 10.8 GPa. This value is just 8% higher than the value measured experimentally (and comparable with the experimental CoV of 7% for the composite modulus).

3. Results

The results from experimental mechanical testing of the fibres (from nine samples) were tensile modulus E_f = 27.7 ± 13.8 GPa, tensile strength σ_f = 430 ± 248 MPa, and elongation at break ε_f = 1.44 ± 0.43%. The coefficients of variation (CoV) for these data were 49.9%, 57.7%, and 29.5%, respectively, confirming the findings of Virk et al. [23] that failure strain is a more reliable design criterion than strength.

The results from experimental mechanical testing of the composites (from four samples with valid failure modes) were E_c = 10.0 ± 0.7 GPa, σ_c = 132 ± 11 MPa, and ε_c = 1.37 ± 0.9%. The interlaminar shear strength (ILSS) experimental results (from 16 samples with valid failure modes) were 14.2 ± 2.5 MPa: a value consistent with those for composites with reasonable fibre/matrix interface bond strength and hence an indication that the assumption of well-bonded fibres is valid for the proposed model.

To determine the fibre volume fraction, samples were prepared for optical microscopy by a coarse polish with 400 grit abrasive to give good contrast. Polishing with fine grit results in a smooth surface free of scratches but having significantly lower contrast. Digital images were obtained using an Olympus BX60M optical microscope (serial number 5M0733) with an Olympus 3030 3.2 megapixel camera, manipulated with Photoshop CS to enhance the contrast so that boundaries could be identified and the image was converted to a binary format (Figure 2). Image analysis was then undertaken using ImageJ to determine the black/white ratio and hence derive the respective fibre volume fractions. Knowledge of the true fibre cross-sectional area allows both κ and η_d to be set to unity. A similar technique was used to determine the void volume fraction.

4. Summary

This paper has presented a novel method for the determination of the density of a natural fibre reinforcement using
Archimedes principle in combination with vacuum degassing of the sample. The new rule of mixtures (ROM) for composites reinforced with non-circular CSA fibres was introduced. ROM has been shown to provide sensible estimate of the elastic modulus of natural fibre composites produced from jute fibres where the fibres have been well characterised. The strain to failure was found to vary less than strength and modulus as previously reported by Virk et al.

Acknowledgments

This paper is based on experimental data in an invited presentation “The determination of the fibre volume fraction in natural fibre composites” at the Exploratory Workshop on Environmentally Friendly Composites (EnviroComp), European Science Foundation Physical and Engineering Sciences Committee, University of Warwick—Coventry, April 20–21, 2004. The same paper was also presented at Universiti Putra Malaysia on September 22, 2004. The authors wish to acknowledge Terry Richards for assistance with mechanical testing and the preparation of samples for optical microscopy and Peter Bond for assistance with image analysis.

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