Dielectric and ferroelectric sensing based on molecular recognition in Cu(1,10-phenlo-throline)₂SeO₄•(diol) systems

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The process of molecular recognition is the assembly of two or more molecules through weak interactions. Information in the process of molecular recognition can be transmitted to us via physical signals, which may find applications in sensing and switching. The conventional signals are mainly limited to light signal. Here, we describe the recognition of diols with Cu(1,10-phenlo-throline)₂SeO₄ and the transduction of discrete recognition events into dielectric and/or ferroelectric signals. We observe that systems of Cu(1,10-phenlo-throline)₂SeO₄•(diol) exhibit significant dielectric and/or ferroelectric dependence on different diol molecules. The compounds including ethane-1,2-diol or propane-1,2-diol just show small temperature-dependent dielectric anomalies and no reversible polarization, while the compound including ethane-1,3-diol shows giant temperature-dependent dielectric anomalies as well as ferroelectric reversible spontaneous polarization. This finding shows that dielectricity and/or ferroelectricity has the potential to be used for signalling molecular recognition.

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Molecular recognition is the weak binding of a guest molecule to a complementary host molecule to form a host-guest complex through non-covalent bonding interactions such as hydrogen bonding, π-π stacking, metal coordination, hydrophobic forces and cation-π interaction. Molecular recognition plays important roles in many biological processes, ranging from enzymatic catalysis, protein synthesis to immunity. Recently, much attention has been directed to the artificial materials and devices based on molecular recognition and related supramolecular chemistry. These host–guest systems can achieve specific functions, such as molecular machines, molecular imprinting, switching, sensing and immunity. The central Cu$^{2+}$ ion has a distorted square-pyramidal coordination geometry defined by four N atoms from two chelating 1,10-phenanthroline ligands and one O atom from a monodentate SeO$_4^{2-}$ anion, and the apex is occupied by a N atom. The monomeric complex and diol molecules are held together by O-H⋯O hydrogen bonding interactions, giving supramolecular structures with an $R_1^2(9)$ ring motif for Cu(1,10-phenanthroline)$_2$SeO$_4$•(ethane-1,2-diol) and Cu(1,10-phenanthroline)$_2$SeO$_4$•(propane-1,2-diol) and an $R_2^2(10)$ ring motif for Cu(1,10-phenanthroline)$_2$SeO$_4$•(propane-1,3-diol). The phase transition temperature of 1 is around 325 K in the cooling run. The high-temperature phase (HTP) structure at 353 K has the centrosymmetric space group C15

**Results**

**Structural phase transition.** Solvate compounds Cu(1,10-phenanthroline)$_2$SeO$_4$•(diol) were obtained as crystals by recrystallization of Cu(1,10-phenanthroline)$_2$SeO$_4$ from diols. They were found to undergo temperature-triggered structural phase transitions by thermal analysis (Supplementary Fig. 1) and dielectric measurements. We determined the temperature-variable crystal structures by X-ray diffraction to understand the origins of the phase transitions and mechanisms of their molecular recognition. These compounds consist of monomeric complex Cu(1,10-phenanthroline)$_2$SeO$_4$ and diol molecules. The central Cu$^{2+}$ ion has a distorted square-pyramidal coordination geometry defined by four N atoms from two chelating 1,10-phenanthroline ligands and one O atom from a monodentate SeO$_4^{2-}$ anion, and the apex is occupied by a N atom. The monomeric complex and diol molecules are held together by O-H⋯O hydrogen bonding interactions, giving supramolecular structures with an $R_1^2(9)$ ring motif for Cu(1,10-phenanthroline)$_2$SeO$_4$•(ethane-1,2-diol) and Cu(1,10-phenanthroline)$_2$SeO$_4$•(propane-1,2-diol) and an $R_2^2(10)$ ring motif for Cu(1,10-phenanthroline)$_2$SeO$_4$•(propane-1,3-diol). The phase transition temperature of 1 is around 325 K in the cooling run. The high-temperature phase (HTP) structure at 353 K has the centrosymmetric space group C15

**Figure 2 | Molecular structures of 1-3.** (a,b) Molecular structures of 1 in the HTP and LTP, respectively. (c,d) Molecular structures of 2 in the HTP and LTP, respectively. The ratios of the two orientations of the SeO$_4^{2-}$ anion are 0.53:0.47 and 0.88:0.12, respectively. (e,f) Molecular structures of 3 in the HTP and LTP, respectively. The temperatures indicate those at which the structures were determined, respectively. The green dashed lines indicate hydrogen bonding interactions. The two orientations of the disordered SeO$_4^{2-}$ anion were distinguished by the two-coloured and the orange bonds. H atoms bonded to the C atoms were omitted for clarity.

**Figure 1 | Structural formula of compounds 1-3.** The dashed lines indicate hydrogen bonding interactions.
Dielectric and ferroelectric properties. The structural analysis reveals the slowing down of dynamics of the SeO$_4^{2−}$ anion with decreasing temperature in 1–3. Such a process is usually accompanied by a dielectric response, which possibly contributes to transmitting the signal of molecular recognition. We measured temperature-variable complex dielectric constant ($\varepsilon = \varepsilon' - i\varepsilon''$), where $\varepsilon'$ is the real part and $\varepsilon''$ is the imaginary part for single crystal samples of the three compounds. As shown in Fig. 5, 1 and 2 show the similar dielectric response, while 3 exhibits distinct dielectric behaviour. For 1 and 2, two anomalous peaks with a temperature gap of about 30–50 K at each measured frequency were observed. It is natural to associate the two anomalies with two structural phase transitions. However, the heat capacity measurements just show one wide thermal anomaly, indicating a single phase transition (Supplementary Fig. 1). The crystal structures determined at intermediate temperatures (303 and 243 K for 1 and 2, respectively) have the same space group as those of the HTPs. Probably, two different polar mechanisms are responsible for the two sequential dielectric anomalies. The anomaly at higher temperature is mild in comparison of those in ferroelectric phase transitions$^{31}$ or in transitions involving rotational dipoles$^{2,33}$, and does not show significant frequency dependence. Such a dielectric response is usually due to the electronic and ionic polarization. The anomaly at lower temperature shows strong frequency dependence in both real and imaginary part (Supplementary Fig. 2). Take 1 as
example, the peak value of $\varepsilon'$ decreases from 95 to 27 and the peaking temperature moves from 280 to 295 K as the frequency increases from 1 kHz to 1 MHz. The low-frequency dispersion is attributable to the dielectric relaxation due to the reorientation of dipoles (SeO$_4^{2-}$ anion). To analyse the relaxation process, the complex dielectric constant of $\varepsilon$ is plotted in Argand diagram and fitted by the Cole–Cole model with the following function $^{34}$:

$$
\varepsilon = \varepsilon(\infty) + \frac{\varepsilon(0) - \varepsilon(\infty)}{1 + (i\omega\tau)^{-h}}
$$

where $\varepsilon(0)$ and $\varepsilon(\infty)$ are the low-frequency and high-frequency values of the real part of dielectric constant, $\tau$ is the relaxation time, $\omega$ is the angular frequency and $h$ is a measure of the distribution of relaxation time. As shown in Fig. 6, the data from 'Cole–Cole arcs' with their centres located below the $\varepsilon'$ axis indicate a polydispersive character. The fitted $h$ values are 0.1835, 0.1628 and 0.1612 at 260, 270 and 280 K, respectively, and $\tau$ values are $5.3 \times 10^{-6}$, $1.8 \times 10^{-6}$ and $6.37 \times 10^{-7}$ s, respectively. A good fit of Cole–Cole model supports that the relaxation process is the reorientation of dipoles, and the low $h$ value indicates a narrow distribution of the relaxation time. The same analysis was also carried out for $\varepsilon''$. The curves deviate significantly from Cole–Cole arcs, indicating a more complex dielectric relaxation process in 1.

For 3, only one $\lambda$-shape anomalous peak appears at each frequency, and the peak heights are significantly larger than those for 1 and 2 in orders of magnitude. The large dielectric constant anomalies reveal the ferroelectric nature of the transition. In the vicinity of the critical temperature, the dielectric response shows Curie–Weiss behaviour, $\varepsilon = C_p(T - T_0)$ ($T > T_c$) or $C_p(T_0 - T)$ ($T < T_c$). The fitted Curie constants at 100 kHz is $C_p = 322$ K and $C_G = 154$ K, and Weiss temperatures $T_0 \approx T_0' = 259.3$ K. The $C_p/C_G$ ratio of 2.09 is quite close to the theoretical value ($C_p/C_G = 2$) expected for a second-order ferroelectric phase transition. The fitted Curie constants at other frequencies are included in Supplementary Table 4.

To identify ferroelectricity, observation of polarization–electric field ($P - E$) hysteresis loops using the Sawyer–Tower circuit is a reliable method$^{31}$. Thus, we examined the $P - E$ dependence of the three compounds (Fig. 7). Compounds 1 and 2 just show the linear dependence at various temperatures, indicating no switchable spontaneous polarization and the lack of ferroelectricity (inset of Fig. 7a). For 3, the polarization response is also linear at temperature above $T_c$, as expected for a paraelectric phase. At a temperature close to $T_c$ (256 K), a flat loop was observed, and a non-zero remnant polarization ($P_r$) at zero field appeared, corresponding to a transition state. Perfect loops were developed at lower temperatures in the stable ferroelectric phase. At 241 K and 50 Hz, we obtained $P_s = 0.70 \mu$C cm$^{-2}$, $P_r = 0.65 \mu$C cm$^{-2}$ and coercive field (the intercept of the loop with the field axis) $E_c = 7.1$ kV cm$^{-1}$. Compared with those in other recently developed molecular ferroelectrics$^{17,18,22,23,27,28,33-43}$, $P_s$ of 3 at 241 K is among the moderate level.

**Discussion**

The ferroelectric mechanism can be interpreted by a combination of structural analysis and theoretical calculation. For compound 3, the molecular electronic dipole moment can be taken as pointing from the Se to Cu atom, since the positive and negative charges are carried mainly by Cu$^{2+}$ ion and SeO$_4^{2-}$ anion, respectively. In the paraelectric phase, the two orientations of SeO$_4^{2-}$ anions were used for the plots. The solid lines represent the best fits using the Cole–Cole model.

![Figure 5](image-url) Dielectric responses of 1–3. (a,e) Temperature dependences of the real part $\varepsilon'$ of complex dielectric constant measured along the $a$ axis at different frequencies for 1–3. (b,f) Temperature dependences of the imaginary part $\varepsilon''$ of complex dielectric constant for 1–3.

![Figure 6](image-url) Argand plots of the complex dielectric constant of 2. The dielectric complex dielectric constants in the temperature range of the dielectric anomalies were used for the plots. The solid lines represent the best fits using the Cole–Cole model.
In the ferroelectric phase, the SeO$_4^{2-}$ anion becomes ordered with a single orientation. The supramolecular in the C$_2$ space group are related by the translations or glides, and thus the dipoles in the $ac$ plane are arranged in parallel, leading to the occurrence of spontaneous polarization (Supplementary Fig. 3). Since the spontaneous polarization in 3 originates from the loss of the C$_2$ axis, the path of polarization reversal can be assumed as rotation of SeO$_4^{2-}$ anion (type A) or SeO$_4^{2-}$-dial as a rigid part (type B) around the (pseudo) C$_2$ axis (Supplementary Fig. 4).

To figure out the detail, we calculated energy barriers for the two rotation types, as shown in Fig. 8. The energy barrier difference of about 80 kJ mol$^{-1}$ indicates that type A is more favourable in 3. This calculation also reveals that the centrosymmetric structure is higher in energy than the ferroelectric one, as expected. Beside these approaches, other possible contributions to the polarization, such as intramolecular charge transfer, are negligible (Supplementary Fig. 5 and Supplementary Table 5).

With this mechanism, we evaluated the crystal polarization by the Berry phase method using a periodic unit cell. The calculated polarization vector of the LTP lies in the $ac$ plane perpendicular to the C$_2$ axis. The vector modulus is 1.71 $\mu$C cm$^{-2}$ and its component along the $a$-direction is 1.36 $\mu$C cm$^{-2}$, which reproduces the experimental value of 0.7 $\mu$C cm$^{-2}$.

The continuous evolution of polarization (both the module and components in the $alc$-direction) from the centrosymmetric ($\lambda = 0$) to the polar structure ($\lambda = 1$) was plotted as a function of dimensionless parameter $\lambda$ in Fig. 7b. The dimensionless parameter $\lambda$ is the normalized amplitude of the atomic displacements in the path from the centrosymmetric structure ($\lambda = 0$) to the polar structure ($\lambda = 1$). Both the rotation of SeO$_4^{2-}$ anion and slight displacement of other atoms are implied in $\lambda$.

The primary feature distinguishing ferroelectrics from other pyroelectrics is that ferroelectric spontaneous polarization can be reversed with an applied electric field$^{23}$. Ferroelectric spontaneous polarization is generated by symmetry breaking, and correspondingly, the crystal structures with the opposite orientation of the polarization are identical or enantiomorphous, and can be transformed into each other by the symmetry operation which is kept just in the paraelectric phase. The two polarization states in 3, for instance, are related by the C$_2$ symmetry. For 2, the dipoles in both the HTP and LTP should be arranged in the same manner as in 3. The ferroelectric polarization reversal requires type B rotation, or the two polarization states will be not symmetrically equivalent, because propane-1,2-diol molecule lacks the C$_2$ symmetry. However, the barrier energy of rotation type B in 2 is 45 kJ mol$^{-1}$ higher than that of rotation type A (Supplementary Fig. 6), indicating that type B rotation is unfavourable and ferroelectric polarization reversal is impossible in the investigated temperature range. The only isosymmetric phase transition in 2 also suggests the difficulty in the polarization reversal. Such polar compounds like 2 are usually regarded as pyroelectrics.

As for 1, the phase transition may involve the type B rotation since the energy barriers for two rotation types are almost equal (Supplementary Fig. 6), different from those in 2 and 3. Although it also undergoes a C$_2$-symmetry-breaking transition, the LTP remains centrosymmetric, and the two orientations of the SeO$_4^{2-}$ anion retain in the crystal with the equal population (Supplementary Fig. 7). Therefore, 1 has no ferroelectric spontaneous polarization.

In summary, diols are recognized by Cu(1,10-phenothroline)$_2$SeO$_4$ through hydrogen bonding interactions to form crystalline compounds with a general formula Cu(1,10-phenothroline)$_2$SeO$_4$·(dial). These compounds exhibit distinct dielectric and/or polar behaviours, depending on the included dial molecules. Both the HTP and LTP of 1 are centrosymmetric,
and thus the crystal shows no polarization. Both the HTP and LTP of 2 are polar, and thus the crystal shows non-switchable polarization. Their phase transitions are accompanied by the moderate dielectric response. Compound 3 has the centrosymmetric HTP and the polar LTP, and thus shows switchable polarization (ferroelectricity) and giant dielectric response. Since the dielectric/ferroelectric properties show high dependence on included diol molecules, the present crystalline compounds offer very attractive perspectives as models of dielectric/ferroelectric sensing. The finding will throw light on the further research on the dielectric/ferroelectric sensing based on molecular recognition, and thus expand the application of molecular ferroelectric materials. From the view point of molecular design, the models can be easily extended to other systems, because the bidentate ligand, diol molecule and the metal ion can be tuned in a wide of range. It is expected that the selenate group can be easily replaced by the sulfate group to maintain the similar structures. Research on the dielectric/ferroelectric sensing properties of these inclusion compounds is in progress.

Methods

Synthesis. 1,10-Phenanthroline (10.0 mmol, 1.80 g), copper(II) carbamate (2.5 mmol, 0.55 g) and selenic acid (40%, 2.00 g) were placed in a 500 ml flask with distilled water (5 ml) and ethanol-1,2-diol (200 ml) as solvents. After refluxing for 4h at 393 K, the solution was cooled to room temperature and then filtered into a 250 ml beaker. Green block crystals of 1 were obtained by slow evaporation of the filtrate at 373 K. Green block crystals of 2 and 3 were prepared using a similar method by replacing ethane-1,2-diol with propane-1,2-diol and propane-1,3-diol, respectively. The purity of the bulk phases was verified by X-ray powder diffraction, infrared and UV–vis spectra (Supplementary Figs 8–10).

Computational details. The crystalline property calculations were performed within the framework of density functional theory (DFT) implemented in the Vienna ab initio Simulation Package. The exchange–correlation interactions were treated within the generalized gradient approximation of the Perdew–Burke–Ernzerhof type. The spontaneous polarization was evaluated by the Berry phase method developed by King-Smith and Vanderbilt. A unit cell with period boundary Ernzerhof type. The spontaneous polarization was evaluated by the Berry phase method, B3LYP-D3, with the Grimme’s DFT-D dispersion correction (Supplementary Fig. 11). The total energy calculations of structures at different (Supplementary Fig. 4). The models of the transition states were obtained by the clockwise or counter-clockwise rotation of half of the SeO\text{2−} anions around the (pseudo) C2 axis. For each transition state (0<\phi<\pi), we took an average of polarization calculated from the models by the clockwise or counter-clockwise rotation (Supplementary Fig. 11).

Energy barrier calculations were carried out with the Gaussian 09 software package. The models structure considering the effects from neighbouring molecules were extracted from the X-ray crystallographically determined geometries (Supplementary Fig. 4). The total energy calculations of structures at different temperatures and rotation angles (0) around the (pseudo) C2 axis were performed by the DFT method, B3LYP-D3, with the Grimme’s DFT-D dispersion correction term, in combination with Stuttgart–Dresden–Bonn relativistic effective core potential SDD used for the Cu atoms (which replaces 10 electrons in inner shells 1 and 2, leaving 17 outer electrons 3p^3\text{3/P}3d^7\text{3/P} as the valence electrons), while the all-electron basis set 6-311G\textsuperscript{**} was applied for Se, S, O, N, C and H atoms (all-electron basis set 6-311G\textsuperscript{**} for N, C and H atoms in the neighbouring molecules). Then, natural population analysis was implemented with NBO (all-electron basis set 6-31G\textsuperscript{**} for N, C and H atoms in the neighbouring molecules). Then, natural population analysis was implemented with NBO (all-electron basis set 6-31G\textsuperscript{**} for N, C and H atoms in the neighbouring molecules).

Data availability. The structures have been deposited at the Cambridge Crystallographic Data Centre (deposition numbers: CCDC 1446837–1446845, 1446843 and 1449519), and can be obtained free of charge from the CCDC via www.ccdc.cam.ac.uk/getstructures.
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Author contributions
W.Q.L. prepared the samples. Y.Z., P.F.L. and D.W.F. characterized the properties. H.Y.Y. determined the structures. Z. N. C, J. Y. W., Q. Z. and J. W performed the calculation. Y.M.Y, S. D. H. and R.G.X. wrote the manuscript. R. G. X. designed and directed the studies.

Additional information
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