Single-phase white-emitting phosphors
Ba$_3$Ce$_{(1-x-y)}$(PO$_4$)$_3$:$x$Tb$^{3+}$,$y$Mn$^{2+}$ and
Ba$_3$Ce$_{(1-x-z)}$(PO$_4$)$_3$:$x$Tb$^{3+}$,$z$Sm$^{3+}$: structure,
luminescence, energy transfer and thermal stability

Shuchao Xu, Zhijun Wang, Panlai Li, Ting Li, Qiongyu Bai, Jiang Sun and Zhiping Yang

A series of Ba$_3$Ce$_{(1-x-y)}$(PO$_4$)$_3$:$x$Tb$^{3+}$,$y$Mn$^{2+}$ and Ba$_3$Ce$_{(1-x-z)}$(PO$_4$)$_3$:$x$Tb$^{3+}$,$z$Sm$^{3+}$ phosphors were synthesized by a high temperature solid-state reaction. X-ray diffraction, luminescence and decay curves were used to characterize the phosphors. All the synthesized phosphors crystallized in the cubic unit cell with $I43d$ space group. Energy can be transferred from Ce$^{3+}$ to Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ in Ba$_3$Ce$_{(1-x)}$(PO$_4$)$_3$:$y$Tb$^{3+}$, Ba$_3$Ce$_{(1-y)}$(PO$_4$)$_3$:$y$Mn$^{2+}$, and Ba$_3$Ce$_{(1-z)}$(PO$_4$)$_3$:$z$Sm$^{3+}$ phosphors. Furthermore, the color of these phosphors can turn from cyan to green, blue to red, and cyan to pale pink. When Ba$_3$Ce(PO$_4$)$_3$ was co-doped with Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$, Tb$^{3+}$ could also transfer part of its energy to Mn$^{2+}$/Sm$^{3+}$, and more importantly, a white emission can be achieved based on this energy transfer. The phosphors exhibited a good thermal stability, with correlated color temperatures of up to 3301 K, and a quantum efficiency as high as 51.2%. These results revealed that we managed to obtain good white emitting phosphors by co-doping Ba$_3$Ce(PO$_4$)$_3$ with Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$.

1 Introduction

The extensive application of white light-emitting diodes (LEDs) has attracted significant attention because of their long lifetime, environmental friendliness, high luminescence efficiency and material stability. Traditional white YAG:Ce$^{3+}$ LEDs were fabricated with a blue-chip, which exhibit high correlated color temperature (CCT $\sim$ 7750 K) and poor color rendering index (CRI 70 $\sim$ 80). Therefore, to obtain high-performance white LEDs, other methods have been introduced. One of the methods for achieving white emission is co-doping different ions in a single-phase compound. This can not only overcome the shortcomings of high correlated color temperature and low color rendering index, but also avoid reabsorption in the host. Generally, energy transfer plays an important role in the optical properties of co-doped luminescent materials, from which we can obtain a tunable color or white emission. Among ions, Ce$^{3+}$ can not only emit blue emission, but can also transfer part of its energy to other ions because of its 5d $\rightarrow$ 4f transition. Moreover, Tb$^{3+}$ ions are widely used as important green activators due to their $^5D_4$, $^7F_j$ transitions ($j =$ 6, 5, 4 and 3). Furthermore, Mn$^{2+}$/Sm$^{3+}$ can be acceptors and can produce the red emission components in the host. Thus, it is possible to obtain a white emission via tri-doping Ce$^{3+}$, Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$ in a single-phase compound.

Among many host compounds, eulytite-type orthophosphates of the M$_1$M$_{II}$M$^{IV}$(PO$_4$)$_4$ ($M^I =$ Ba, Sr, Ca, and Pb, $M^{II} =$ La, Sc, Y, Bi, and In) type have attracted significant interest for their good optical properties and thermal stability. Ba$_3$Ce(PO$_4$)$_3$ (BCP) is a eulytite-type phosphate, with the advantage of having Ce$^{3+}$ as one of the matrix cations. Thus, the doping concentration of Ce$^{3+}$ is up to 100%, such that Ce$^{3+}$ can transfer a large amount of energy to the co-doping ions in the host. Therefore, in our study, a series of white-emitting Ba$_3$Ce(PO$_4$)$_3$:Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ phosphors were investigated, along with the energy transfer from Ce$^{3+}$ to other activators. Importantly, the white emitting phosphors Ba$_3$Ce(PO$_4$)$_3$:Tb$^{3+}$/Mn$^{2+}$ and Ba$_3$Ce(PO$_4$)$_3$:Tb$^{3+}$/Sm$^{3+}$ exhibited a good thermal stability and correlated color temperatures.

2 Experimental

2.1 Materials and synthesis

A series of Ba$_3$Ce$_{1-x-y}$PO$_4$:xTb$^{3+}$,yMn$^{2+}$ and Ba$_3$Ce$_{1-x-z}$PO$_4$:xTb$^{3+}$,zSm$^{3+}$ phosphors were prepared by a high-temperature solid-state reaction process. The raw materials BaCO$_3$ (analytical reagents, A.R.), NH$_4$H$_2$PO$_4$ (A.R.), CeO$_2$ (99.99%), Tb$_2$O$_7$ (99.99%), Sm$_2$O$_3$ (99.99%) and MnCO$_3$ (A.R.) were weighed using an electronic scale with a 0.0001 g accuracy. The stoichiometric amounts of raw materials were thoroughly mixed.
mixed and ground using an agate mortar and pestle for more than 30 min until they became uniformly distributed. Then, the obtained material was transferred into an alumina crucible and sintered at 1150 °C for 4 h in a reductive atmosphere (20% H₂ + 80% N₂). After sintering, samples were cooled down to room temperature and then ground again into a powder for subsequent use.

2.2 Material characterization
X-ray diffraction (XRD) patterns of the as-synthesized samples were obtained to determine the crystal structure, using a D/max-ra with Cu Kα radiation at 40 kV and 40 mA. XRD patterns, which were analyzed by Rietveld refinement, were acquired at the step size of 0.05° with a counting time of 2 s per step. Findit software and Crystalmaker software were used for identifying the structure. Spectral properties were analyzed via a F-4600 spectrophotometer equipped with a 450 W Xe lamp. Temperature-dependent luminescence properties were measured via the same spectrophotometer, which was assembled with a TAP-02 high-temperature fluorescence controller. The Commission International de l’Eclairage (CIE) chromaticity coordinates of the samples were measured by a PMS-80 UV-VIS-NEAR IR spectra analysis system. Decay curves of Tb³⁺/Sm³⁺/Mn²⁺ were obtained using a 450 W Xe lamp as the excitation source (HORIBA, FL-1057). Decay curves of Ce³⁺ were obtained using a nona LED (370 nm) as the excitation source. Furthermore, particle morphology was investigated by scanning electron microscopy (SEM), and images were obtained using Nova NanoSEM 650 at the accelerating voltage of 10 kV.

3 Results and discussion
3.1 Phase formation
The phase purity of our phosphors was evaluated by X-ray diffraction. Fig. 1 exhibits the XRD patterns of representative phosphors as well as the standard card of Ba₃La(PO₄)₃ for comparison. It can be found that all the diffraction peaks of the samples can be exactly assigned to the pure cubic phase Ba₃La(PO₄)₃ (JCPDS 85-2448). No other phase or impurity were detected, indicating that the doped ions cannot induce significant changes in the crystal structure. In other words, our samples were obtained without any heterogeneous phase. The SEM images of BCP, BCP:0.07Tb³⁺, BCP:0.07Tb³⁺,0.07Mn²⁺ and BCP:0.07Tb³⁺,0.07Sm³⁺ are shown in the lower part of Fig. 1. It was found that the samples consisted of aggregated particles with a sphere-like morphology of an average size of 5–10 μm. A micrometer-size of the phosphor particles is essential for high luminescence, and the synthesized phosphors exhibit a particle size in the micrometer range, and hence, they can exhibit a good luminescence.

For the purpose of further elucidating the crystal structure of the samples, we analyzed the XRD patterns by Rietveld refinements, as represented in Fig. 2(a–e). The multiplication sign denotes the XRD pattern of the samples, the red lines represent the corresponding Rietveld refinement, and the blue line represents the residuals. The refinement results, as presented in Table 1, further demonstrate that the phosphors are single-phase without any impurity or secondary phases. As shown in Fig. 2(f), (I) is the crystal structure of Ba₃La(PO₄)₃ in a cubic system with space group I43d of the unit cell, and (II) and (III) are the coordination environments of the P⁵⁺ and Ba²⁺/La³⁺/Ce³⁺ sites. In the structure, each P⁵⁺ is surrounded by 8 O²⁻ anions to form an octahedron with four long P–O distances of 1.501 Å and four short P–O distances of 1.488 Å. Each Ba²⁺ is coordinated with 18 non-equidistant oxygens at the distance ranging from 2.704 Å to 2.973 Å.

3.2 Luminescence properties
The inset of Fig. 3 shows the emission and excitation spectra of BCP. Under excitation at 325 nm, BCP shows a broad-band blue emission ranging from 330 to 550 nm, centered at 387 nm. This emission is attributed to the transition of Ce³⁺ from the 5d excited state to the 4f ground state. The corresponding excitation spectrum (monitored at 387 nm) covering the range of 200–350 nm is attributed to the 4f → 5d transition of Ce³⁺. Fig. 3 shows the decay curve of BCP, and the luminescence decay time can be calculated by the following formula:  

\[ \tau = \frac{\left( \int I(t)dt \right)}{\left( \int I(t)dt \right)} \]  

where \( I(t) \) represents the luminescence intensity at a time \( t \), \( \tau \) is the time, and \( \tau \) is the lifetime. The calculated decay time is...
shown in Fig. 3. Moreover, the measured quantum efficiency of the host was 55.3%.

To study whether there is energy transfer from Ce to Tb, we studied the emission and excitation spectra of BCP and BTP (Ba₃Tb(PO₄)₃), as shown in Fig. 4. For BTP, the excitation spectrum monitored at 551 nm contains two broad bands and many other bands, which are attributed to f/d and f/f transitions in Tb³⁺. The emission spectrum at 377 nm excitation consists of several bands from 480 to 650 nm, attributed to 5D⁴/7F J (J = 6, 5, 4, 3) transitions in Tb³⁺. Moreover, the results show that there is an overlap between the emission spectrum of BCP and the excitation spectrum of BTP. Therefore, we hypothesized that Ce³⁺ might transfer its energy to Tb³⁺ in eulytite-type phosphates. Fig. 4(c) and (d) depict the emission and excitation spectra of BCP:0.03Tb³⁺ and BTP:0.03Ce³⁺, respectively. It can be clearly seen that the excitation spectra monitored at 551 nm for BCP:0.03Tb³⁺ is similar to the excitation spectrum monitored at 387 nm for BCP. Moreover, both emission spectra (λex = 325 nm) of the doped BCP and BTP display the characteristic peaks of Ce³⁺ and Tb³⁺. Thus, we synthesized BCP:xTb³⁺ to further explore whether energy transfer occurs. As depicted in Fig. 4(e), the emission spectra of all samples contain both Ce³⁺ and Tb³⁺ emission bands. In addition, as shown in Fig. 4(f), the emission intensity of Ce³⁺ monotonously decreases as the Tb³⁺ concentration increases, which evidences the occurrence of energy transfer from Ce³⁺ to Tb³⁺. Importantly, the emission color shifts from cyan to green as Tb³⁺ concentration increases in BCP:xTb³⁺.

To obtain white emission, there should be a red component in the emission spectrum of the phosphor. Thus, we first synthesized a series of Ce³⁺/Mn²⁺ and Ce³⁺/Sm³⁺ co-doped phosphors. To investigate the luminescence characteristics, the emission and excitation spectra were obtained and are shown in Fig. 5(a). The excitation spectra of BCP:0.10Sm monitored at 606 nm and 387 nm are similar in the range from 200 to 370 nm. The differences from 370 to 550 nm is due to the transitions of Sm³⁺. Furthermore, the excitation spectrum monitored at 606 nm has a broad overlap with the emission spectrum excited at 325 nm of BCP. Thus, energy transfer from Ce³⁺ to Sm³⁺ ions in BCP:0.10Sm³⁺ may take place. Moreover, we synthesized a series of BCP:Sm and their spectra at 325 nm excitation contain both Ce³⁺ and Sm³⁺ emission bands, as shown in Fig. 5(b). The inset is the enlarged spectra from 560 to 680 nm centered at 565, 606 and 650 nm, which are assigned to the 4G⁵/₂/6H⁷/₂ and 6H⁹/₂ transitions of Sm³⁺, respectively. As depicted in Fig. 5(c), the emission intensity of Ce³⁺ decreases as the Sm³⁺ concentration increases in BCP:0.10Sm³⁺.

Table 1 XRD refinement results of the samples

|                  | BCP:0.07Tb | BCP:0.07Mn | BCP:0.10Sm | BCP:0.07Tb,0.07Mn | BCP:0.07Tb,0.07Sm |
|------------------|------------|------------|------------|-------------------|-------------------|
| χ²               | 2.850      | 1.704      | 2.058      | 1.983             | 1.587             |
| R_p              | 8.49%      | 7.48%      | 8.44%      | 8.04%             | 7.30%             |
| R_wp             | 12.67%     | 7.84%      | 10.18%     | 10.16%            | 7.15%             |
| a = b = c        | 10.500     | 10.512     | 10.520     | 10.501            | 10.514            |
| Cell volume/A³   | 1157.158   | 1161.657   | 1164.305   | 1158.073          | 1162.349          |
| α = β = γ        | 90°        | 90°        | 90°        | 90°               | 90°               |
| Z                 | 4          | 4          | 4          | 4                 | 4                 |
| Space group      | I43d       | I43d       | I43d       | I43d              | I43d              |
concentration increases. Moreover, the emission intensity of Sm$^{3+}$ steeply increases until it reaches a maximum at $z = 0.10$. Then, due to concentration quenching, the emission intensity gradually decreases, which further evidences the energy transfer from Ce$^{3+}$ to Sm$^{3+}$ ions. As a result of the energy transfer, in the co-doped BCP:zSm$^{3+}$, the color can turn from cyan to pale pink.

In addition, as shown in Fig. 5(d), the excitation spectra of Mn$^{2+}$ ion monitored at 606 nm is exactly the same to that of Ce$^{3+}$ ion monitored at 387 nm in BCP:0.07Mn$^{2+}$. This implies that energy transfer from Ce$^{3+}$ to Mn$^{2+}$ ions should also occur. To further prove this hypothesis, the emission spectra of BCP:zMn$^{2+}$ at 325 nm excitation were obtained and are shown in Fig. 5(e). There are two broad emission bands, the one from 350 to 500 nm is attributed to the 5d to 4f transition in Ce$^{3+}$, and the other band, from 550 nm to 700 nm is attributed to the $^{4}T_{1}(^{6}G) \rightarrow ^{6}A_{1}(^{6}S)$ spin-forbidden transition of Mn$^{2+}$.\textsuperscript{28,29}

The variation in emission intensity with different Mn$^{2+}$ contents, depicted in Fig. 5(f), shows that the intensity of Ce$^{3+}$ gradually decreases, whereas the intensity of Mn$^{2+}$ first increases, and then decreases as the concentration of Mn$^{2+}$ increases due to concentration quenching. Thus, these results evidence the occurrence of energy transfer from Ce$^{3+}$ to Mn$^{2+}$ ions. As a result of the energy transfer, in the co-doped BCP:yMn$^{2+}$, the color can change from blue to red.
concentration and the other is the energy transfer from Ce to the doping ions. However, when the concentration of doping ions is high enough, concentration quenching occurs. Thus, the results once again confirm the energy transfer from Ce$^{3+}$ to Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ in BCP:xTb$^{3+}$, BCP:yMn$^{2+}$ and BCP:zMn$^{3+}$.

The energy transfer efficiency (E) was calculated from the decay lifetimes by the following equation:

$$\eta = 1 - \tau / \tau_0$$  \hspace{1cm} (2)

where $\tau$ and $\tau_0$ are the lifetimes of sensitizer ions with and without the presence of an activator, respectively. The energy transfer efficiencies are also shown in Table 2.

For investigating the energy transfer mechanism from Ce$^{3+}$ to Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ in BCP:xTb$^{3+}$, BCP:yMn$^{2+}$ and BCP:zMn$^{3+}$, the critical distance ($R_c$) between sensitizer and activator needs to be calculated. According to the Dexter’s theory,\textsuperscript{a0} there are two types of interactions between sensitizer and activator: exchange interaction and multipolar interaction. The distance between sensitizer and activator becomes increasingly shorter as the concentration of activator increases. Thus, the probability of energy transfer will increase, which will cause the concentration quenching to occur and the energy migration should be hindered. The critical distance for energy transfer between sensitizer and activator in the phosphors can be calculated by equation:\textsuperscript{a4}

$$R_c = 2(3V(4\pi XZ))^{1/3}$$  \hspace{1cm} (3)

where $V$ is the volume of the unit cell, $Z$ is the number of host cations per unit cell, and $X$ is the total concentration of sensitizer and activator at which the emission intensity of Ce$^{3+}$ is half the emission intensity of the sample in the absence of Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$. For BCP, V is 1163.33 Å$^3$, Z is 4. Moreover, $X$ are 0.10, 0.07, and 0.03 for BCP:xTb$^{3+}$, BCP:yMn$^{2+}$ and BCP:zMn$^{3+}$, respectively. Thus, the corresponding critical distances ($R_c$) were estimated to be about 17.7 Å, 19.8 Å, and 26.46 Å for BCP:xTb$^{3+}$, BCP:yMn$^{2+}$ and BCP:zMn$^{3+}$, respectively. These results indicate that exchange interaction is almost impossible as exchange interaction is predominant only for about 5 Å. Therefore, the energy transfer mechanism should occur through electric multipolar interaction. According to the Dexter’s multipolar interaction formula and the Reisfeld’s approximation, the following relationship can be applied:\textsuperscript{a2,a3}

$$I_{a0}/I_a \propto C^{2/3}$$  \hspace{1cm} (4)

$$I_{a0}/I_a \propto C^{2/3}$$  \hspace{1cm} (5)

where $I_{a0}$ and $I_a$ are the luminescence quantum efficiency of Ce$^{3+}$ ions without and with Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ ions, respectively. $C$ is the total concentration of Ce$^{3+}$ and Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ ions. The value of $\alpha$ can be 6, 8, and 10, corresponding to dipole–dipole, dipole–quadrupole, and quadrupole–quadrupole interactions, respectively. In addition, the ratio of quantum efficiencies can be calculated from the ratio of emission intensity by the following expression:\textsuperscript{a4,a5}

$$I_{a0}/I_a \propto C^{2/3}$$

### Table 2: Decay time values of Ce/Tb/Mn/Sm in the phosphors and efficiency of the energy transfer from Ce to the doping ions

| Doping Ions | $\tau_{Ce}$ (ns) | $\tau_{Ce}$/$\tau_{Mn}$/$\tau_{Sm}$ (ms) | $\eta$ |
|-------------|------------------|--------------------------------------|-------|
| BCP:xTb     |                  |                                      |       |
| $x = 0.03$  | 10.311           | 3.494 0.534                          |       |
| $x = 0.07$  | 10.300           | 3.587 0.532                          |       |
| $x = 0.10$  | 8.809            | 3.499 0.456                          |       |
| $x = 0.13$  | 8.296            | 3.398 0.429                          |       |
| $x = 0.15$  | 7.546            | 3.296 0.390                          |       |
| $x = 0.20$  | 5.020            | 3.180 0.260                          |       |
| BCP:yMn     |                  |                                      |       |
| $y = 0.03$  | 8.393            | 26.451 0.434                         |       |
| $y = 0.07$  | 8.833            | 30.531 0.356                         |       |
| $y = 0.10$  | 6.383            | 29.044 0.330                         |       |
| $y = 0.13$  | 6.316            | 25.772 0.327                         |       |
| $y = 0.15$  | 6.287            | 25.089 0.325                         |       |
| $y = 0.20$  | 6.176            | 22.010 0.319                         |       |
| BCP:zMn     |                  |                                      |       |
| $z = 0.03$  | 15.610           | 28.762 0.807                         |       |
| $z = 0.07$  | 12.051           | 28.958 0.612                         |       |
| $z = 0.10$  | 11.827           | 28.909 0.623                         |       |
| $z = 0.13$  | 9.291            | 22.058 0.513                         |       |
| $z = 0.15$  | 9.496            | 16.233 0.491                         |       |
| $z = 0.20$  | 9.197            | 15.650 0.476                         |       |
doping Tb³⁺ and Mn²⁺/Sm³⁺ ions in BCP may render white light. Fig. 8(A) shows the emission spectra of BCP:0.07Tb³⁺,yMn²⁺. It can be seen that the characteristic peaks of Ce³⁺, Tb³⁺, and Mn²⁺ all appear in the spectra. Moreover, the intensity of the peaks vary with the concentration of Mn²⁺. The inset shows the variation trend for the intensity. As the concentration of Mn²⁺ increases, the emission intensity of Ce³⁺ and Tb³⁺ all steadily decrease, whereas the emission intensity of Mn²⁺ increases at the beginning and then decreases. The decrease in the emission of Ce³⁺ is due to the energy transfer from Ce³⁺ to Tb³⁺ and Mn²⁺, and the variation in Mn²⁺ emission is a result of concentration quenching. Moreover, the concentration quenching point for Mn²⁺ is 0.05, which is lower than that in BCP:zMn. Thus, we considered that there may be an additional energy transfer to Mn²⁺, as well as from Ce³⁺. Moreover, the emission intensity of Tb³⁺ gradually decreases although there is energy transfer from Ce³⁺ to Tb³⁺. All these results indicate that energy transfer may also occur from Tb³⁺ to Mn²⁺/Sm³⁺ in BCP:0.07Tb³⁺,yMn²⁺/Sm³⁺. Similarly, the comparison of Sm³⁺ concentration quenching point between BCP:0.07Tb³⁺,zSm³⁺ and BCP:zSm³⁺, and the decreased emission of Tb³⁺ indicate energy transfer from Tb³⁺ to Sm³⁺ in BCP:0.07Tb³⁺,zSm³⁺.

To further investigate the energy transfer from Tb³⁺ to Mn²⁺/Sm³⁺ in BCP:0.07Tb³⁺,yMn²⁺ and BCP:0.07Tb³⁺,zSm³⁺, the decay curves of Tb³⁺ (λ_em = 551 nm, λ_ex = 377 nm), Mn²⁺ (λ_em = 606 nm, λ_ex = 387 nm), and Sm³⁺ (λ_em = 606 nm, λ_ex = 387 nm) were obtained and are plotted in Fig. 9. The corresponding decay time values, calculated by formula (1), are listed in Table 3. Upon comparing the decay time values in Table 3 with the values in Table 2, we found significant differences. First, the decay time of Tb³⁺ in BCP:0.07Tb³⁺ is 3.387 ms. On the other hand, the decay time of Tb³⁺ in both BCP:0.07Tb³⁺,yMn²⁺ and BCP:0.07Tb³⁺,zSm³⁺ decreases as the concentration of doping ion increases. Moreover, we found that the decay time values of Mn²⁺ in BCP:0.07Tb³⁺,yMn²⁺ are also greater than those in BCP:zMn at the same concentration of Mn²⁺, which is also the same for the decay time values of Sm³⁺. Based on these results, we can confirm that there is energy transfer from Tb³⁺ to Mn²⁺/Sm³⁺. Moreover, the decay time of Mn²⁺/Sm³⁺ all increase at the beginning, reach a maximum at the concentration of 0.07, and thereafter decrease, which is consistent with the variation in emission intensity in Fig. 8. The change in the Mn²⁺/Sm³⁺ decay time is a combination of energy transfer from Ce³⁺/Tb³⁺ to the doping ions and concentration quenching. Moreover, we also calculated the decay time values of Ce³⁺ (λ_em = 325 nm, λ_ex = 387 nm) in BCP:0.07Tb³⁺,yMn²⁺ and BCP:0.07Tb³⁺,zSm³⁺, and found that the values are all smaller than in BCP:zTb³⁺, BCP:zMn²⁺, or BCP:zSm³⁺ at the same concentration of doping ions, further verifying the energy transfer from Ce³⁺ to Tb³⁺/Mn²⁺/Sm³⁺.
relationship among the ions in BCP:0.07Tb$_{3+}$,yMn$_{2+}$ and BCP:0.07Tb$_{3+}$,zSm$_{3+}$.

3.3 CIE coordinates

The CIE chromaticity coordinates and the corresponding images of BCP:0.07Tb$_{3+}$,yMn$_{2+}$ and BCP:0.07Tb$_{3+}$,zSm$_{3+}$ are shown in Fig. 11, along with the images of BCP:xTb$_{3+}$, BCP:yMn$_{2+}$ and BCP:zSm$_{3+}$. The emitting colour can turn from blue to red by adjusting the concentration of Mn$_{2+}$ in BCP:yMn$_{2+}$. Moreover, the colour of BCP:zSm$_{3+}$ can change from cyan to pale pink, and the colour can vary from cyan to green in BCP:xTb$_{3+}$. In other words, we obtained a series of different emission colours while co-doping the host with ions at different concentrations. More importantly, the CIE chromaticity coordinates (x, y) for BCP:0.07Tb$_{3+}$,yMn$_{2+}$ excited at 365 nm vary from (0.2200, 0.3010) to (0.5143, 0.3163), with white emission of BCP:0.07Tb$_{3+}$,0.07Mn$_{2+}$. Moreover, the CIE chromaticity coordinates (x, y) for BCP:0.07Tb$_{3+}$,zSm$_{3+}$ vary from (0.1995, 0.2893) to (0.4187, 0.2302), with white emission of BCP:0.07Tb$_{3+}$,0.05/0.07Sm$_{3+}$. Therefore, we obtained white emission via co-doping Tb$_{3+}$ and Mn$_{2+}$/Sm$_{3+}$ in BCP. Additionally, the correlated colour temperatures of BCP:0.07Tb$_{3+}$,0.07Mn$_{2+}$, and BCP:0.07Tb$_{3+}$,0.05/0.07Sm$_{3+}$ were determined to be 3812, 3196, and 3301 K, respectively, with a quantum efficiency of up to 47.9%, 48.6%, and 51.2%, respectively.

3.4 Thermal quenching properties

Thermal stability is an important parameter for phosphors as it can affect the color rendering and emission intensity. To investigate the effect of temperature on luminescence properties, the temperature-dependent emission spectra of BCP:0.07Tb$_{3+}$,0.07Mn and BCP:0.07Tb$_{3+}$,0.07Sm were obtained and are shown in Fig. 12(a) and (b), where the insets show that the emission intensity depends on the temperature. It is obvious

![Fig. 9](image-url) (a–c) Decay curves of Ce$_{3+}$, Tb$_{3+}$, and Mn$_{2+}$ in BCP:0.07Tb$_{3+}$,yMn$_{2+}$; (d–f) decay curves of Ce$_{3+}$, Tb$_{3+}$, and Sm$_{3+}$ in BCP:0.07Tb$_{3+}$,zSm$_{3+}$.

![Fig. 10](image-url) Schematic of the energy transfer relationship among the ions in the phosphors.

|     | $\tau_{\text{Ce}}$ (ns) | $\tau_{\text{Tb}}$ (ms) | $\tau_{\text{Mn/Sm}}$ (ms) |
|-----|------------------------|------------------------|------------------------|
| BCP:0.07Tb$_{3+}$,yMn$_{2+}$ | y = 0.03 | 4.505 | 1.455 | 34.666 |
|     | y = 0.07 | 4.330 | 1.334 | 36.495 |
|     | y = 0.10 | 4.155 | 1.153 | 35.344 |
|     | y = 0.13 | 3.757 | 1.126 | 33.073 |
|     | y = 0.15 | 3.509 | 0.777 | 32.372 |
|     | y = 0.20 | 3.406 | 0.576 | 25.810 |
| BCP:0.07Tb$_{3+}$,zSm$_{3+}$ | z = 0.03 | 8.615 | 2.656 | 48.238 |
|     | z = 0.07 | 6.800 | 2.173 | 54.133 |
|     | z = 0.10 | 6.028 | 2.139 | 46.310 |
|     | z = 0.13 | 5.523 | 2.043 | 41.372 |
|     | z = 0.15 | 4.481 | 2.001 | 40.535 |
|     | z = 0.20 | 3.878 | 1.912 | 40.337 |
that the emission of Ce$^{3+}$ in BCP:0.07Tb$^{3+}$,0.07Mn$^{2+}$ and BCP:0.07Tb$^{3+}$,0.07Sm$^{3+}$ decreases as the temperature increases. By contrast, the emission intensity of Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$ in the phosphors first increases up to a maximum and then decreases. Generally, the emission intensity decreases as temperature increases due to the increase in thermal activation through the crossing point between the ground state and the excited state.$^{36,37}$ In our study, the increase in Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$ emission intensity may be the result of energy transfer from Ce$^{3+}$ to the other ions. In other words, the energy transfer efficiency from Ce$^{3+}$ to the doping ions increases with temperature.$^{38}$ Fig. 12(e) shows the temperature-dependent emission spectra of BCP, where the emission intensity also decreases with temperature. Moreover, we compared the emission intensity of Ce$^{3+}$ in BCP, BCP:0.07Tb,0.07Mn and BCP:0.07Tb,0.07Sm at various temperatures, as shown in Fig. 12(f). We can see that the emission intensity of Ce$^{3+}$ in BCP is higher than that in the co-doped samples at all temperatures, which is due to energy transfer from Ce$^{3+}$ to the doping ions. The results suggest that Ce$^{3+}$ may transfer more energy to the ions in the host at higher temperatures. The activation energy for thermal quenching was calculated by the Arrhenius equation:$^{39,40}$

$$\ln(I/I_0) = - E_a/RT$$

where $I_0$ and $I$ are the emission intensity of the phosphors at room temperature and at the experimental temperature, respectively; $K$ is the Boltzmann constant ($8.626 \times 10^{-5}$ eV); $A$ is a constant; and $E_a$ is the activation energy for thermal quenching. Based on the abovementioned equation, the fitting curve of $\ln((I/I_0) – 1)$ versus $1/KT$ was drawn and illustrated in Fig. 12(c) and (d). The calculated values of $E_a$ were determined to be 0.2595 eV and 0.2346 eV for BCP:0.07Tb$^{3+}$,0.07Mn$^{2+}$ and BCP:0.07Tb$^{3+}$,0.07Sm$^{3+}$, respectively, which are higher than those for YAG:Ce$^{3+}$ ($E_a = 0.136$ eV).$^{44}$

4 Conclusions

In summary, we synthesized a series of BCP:xCe$^{3+}$, BCP:xMn$^{2+}$, and BCP:xSm$^{3+}$ phosphors and investigated their luminescence properties. The energy transfer from Ce$^{3+}$ to Tb$^{3+}$/Mn$^{2+}$/Sm$^{3+}$ occurs in BCP:xCe$^{3+}$, BCP:xMn$^{2+}$, and BCP:xSm$^{3+}$ via quadrupole–quadrupole, dipole–dipole, and dipole–dipole interactions, respectively, with a tunable color. More importantly, when co-doped Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$ in BCP, the energy of Ce$^{3+}$ can not only transfer to Tb, but also to Mn/Sm$^{3+}$. Moreover, Tb$^{3+}$ can also transfer part of its energy to Mn$^{2+}$/Sm$^{3+}$ in the phosphors. Above all, we obtained a white emission based on the energy transfer among the ions, and the correlated color temperatures can reach up to about 3200 K, and the quantum efficiency can be as high as 52%. In summary, the results show that we managed to obtain white phosphors by co-doping a Ba$_2$Ce(PO$_4$)$_3$ host with Tb$^{3+}$ and Mn$^{2+}$/Sm$^{3+}$.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066), the Funds for Distinguished Young Scientists of the Hebei Province, China (No. A2015201129), the Natural Science Foundation of China (No. 50902042; 51672066).
personnel training project of the Hebei Province, China (No. A2016002013), and the Post-graduate’s Innovation Fund Project of the Hebei University (No. X2016064, X2016063).

Notes and references

1 P. F. Smet, A. B. Parmentier and D. Poelman, J. Electrochem. Soc., 2011, 158, R37–R54.
2 C. C. Lin and R. S. Liu, J. Phys. Chem. Solids, Lett. Sect., 2011, 2(11), 1268–1277.
3 L. Chen, C. C. Lin, C. W. Yeh and R. S. Liu, Materials, 2010, 3(3), 2172–2195.
4 N. C. George, K. A. Denault and R. Seshadri, Annu. Rev. Mater. Res., 2013, 43, 481–501.
5 Z. G. Xia and R. S. Liu, J. Phys. Chem. C, 2012, 116, 15604–15609.
6 G. Y. Lee, J. Y. Han, W. B. Im, S. H. Cheong and D. Y. Jeon, Inorg. Chem., 2012, 51(20), 10688–10694.
7 H. S. Roh, S. Hur, H. J. Song, I. J. Park, D. K. Yim, D. W. Kim and K. S. Hong, Mater. Lett., 2012, 70(3), 37–39.
8 V. Bachmann, C. Ronda and A. Meijerink, Chem. Mater., 2009, 21, 2077–2084.
9 S. Ye, F. Xiao, Y. X. Pan, Y. Y. Ma and Q. Y. Zhang, Mater. Sci. Eng., R, 2010, 71(1), 1–34.
10 W. R. Liu, C. H. Huang, C. W. Yeh, J. C. Tsai, Y. C. Chiu, Y. T. Yeh and R. S. Liu, Inorg. Chem., 2012, 51, 9636–9641.
11 R. Yu, H. M. Noh, B. K. Moon, C. C. Byung, H. J. Jung, J. Kiwan, S. Y. Soung and K. J. Jun., J. Alloys Compd., 2013, 576(29), 236–241.
12 C. C. Zhao, X. Yin, Y. M. Wang, F. Q. Huang and Y. H. Jang, J. Lumin., 2012, 132(3), 617–621.
13 W. Zhou, D. Hou, F. Pan, B. Zhang, P. Dorenbos, Y. Huang, Y. Tao and H. Liang, J. Mater. Chem. C, 2015, 3, 9161–9169.
14 J. He, R. Shi, M. G. Brik, P. Dorenbos, Y. Huang, Y. Tao and H. Liang, J. Lumin., 2015, 161, 257–263.
15 D. Hou, C. Liu, X. Kuang and H. Liang, Opt. Express, 2012, 20(27), 28969–28980.
16 L. Zhou, H. Liang, P. A. Tanner, S. Zhang, D. Hou, C. Liu and L. Li, J. Mater. Chem. C, 2013, 1(43), 7155–7165.
17 T. Sheng, Z. Fu, X. Wang, S. Zhou, S. Zhang and Z. Dai, J. Phys. Chem. C, 2012, 116(36), 19597–19603.
18 C. H. Huang, L. Y. Luo and T. M. Chen, J. Electrochem. Soc., 2011, 158, J341–J344.
19 C. H. Hsu and C. H. Lu, J. Mater. Chem., 2011, 21(9), 2932–2939.
20 T. Li, P. L. Li, Z. J. Wang, S. C. Xu, Q. Y. Bai and Z. P. Yang, RSC Adv., 2015, 5, 71735–71742.
21 N. Guo, Y. Jia, W. Lu, W. Lv, Q. Zhao, M. Jiao, B. Shao and H. You, Dalton Trans., 2013, 42, 5649–5654.
22 Z. Wang, S. Lou and P. Li, J. Lumin., 2014, 156, 87–90.
23 N. Guo, Y. Huang, Y. Jia, W. Lv, Q. Zhao, W. Lü, Z. Xia and H. You, Dalton Trans., 2013, 42(4), 941–947.
24 H. P. Ji, Z. H. Huang, Z. G. Xia, M. S. Molokeev, X. X. Jiang, Z. S. Lin and V. V. Atuchin, Dalton Trans., 2015, 44, 7679–7686.
25 T. Li, P. L. Li, Z. J. Wang, S. C. Xu, Q. Y. Bai and Z. P. Yang, Inorg. Chem., 2016, 55, 8758–8769.
26 C. Wang, P. L. Li, Z. J. Wang, Y. S. Sun, J. G. Cheng, Z. L. Li, M. M. Tian and Z. P. Yang, Phys. Chem. Chem. Phys., 2016, 18, 28661–28673.
27 Y. Jia, H. Qiao, Y. Zheng, N. Guo and H. You, Phys. Chem. Chem. Phys., 2012, 14, 3337–3342.
28 N. Guo, H. P. You, Y. H. Song, M. Yang, K. Liu, Y. H. Zheng, Y. J. Huang and H. J. Zhang, J. Mater. Chem., 2010, 20, 9061–9067.
29 F. Xiao, E. H. Song and Q. Y. Zhang, Spectrochim. Acta, Part A, 2014, 122, 343–347.
30 D. L. Dexter and J. H. Schulman, J. Chem. Phys., 1954, 22(6), 1063–1070.
31 D. L. Dexter, J. Chem. Phys., 1953, 21(5), 836–850.
32 R. Reisfeld, E. Greenberg, R. Velapolbi and B. Barnett, J. Chem. Phys., 1972, 56(4), 1698–1705.
33 C. H. Huang, T. W. Kuo and T. M. Chen, ACS Appl. Mater. Interfaces, 2010, 2(5), 1395–1399.
34 K. H. Kwon, W. B. Im, H. S. Jang, H. S. Yoo and D. Y. Jeon, Inorg. Chem., 2009, 48(11), 11525–11532.
35 P. Paulose, G. Jose, V. Thomas, N. Unnikrishnan and M. Warrier, J. Phys. Chem. Solids, 2003, 64(5), 841–846.
36 K. Li, M. J. Xu, J. Fan, M. M. Shang, H. Z. Lian and J. Lin, J. Mater. Chem. C, 2015, 3, 11618–11628.
37 Y. P. Varshini, Physica, 1967, 34(1), 149–154.
38 C. Wang, P. L. Li, Z. J. Wang, Y. S. Sun, J. G. Cheng, Z. L. Li, M. M. Tian and Z. P. Yang, Phys. Chem. Chem. Phys., 2016, 18, 28661–28673.
39 W. Lv, Y. Jia, Q. Zhao, M. Jiao, B. Shao, W. Lü and H. You, Adv. Opt. Mater., 2013, 2(2), 183–188.
40 Y. H. Chen, B. Liu, C. S. Shi, G. H. Ren and G. Zimmerer, Nucl. Instrum. Methods Phys. Res., 2005, 537(1–2), 31–35.
41 Y. F. Zhang, L. Li, X. S. Zhang and Q. Xi, J. Rare Earths, 2008, 26, 446–449.