Influence of low temperature annealing on Nickel RTDs designed for heat flux sensing

Youssef Mokadem¹, Sylvie Bégot¹*, François Lanzetta¹, Jean-Yves Rauch¹, Aymen Zribi¹, Virginie Moutarlier², Magali Barthès¹

¹FEMTO-ST Institute UMR 6174, CNRS-University Bourgogne Franche-Comté-ENSMM-UTBM, 2 avenue Jean Moulin, 90000 Belfort France
²UTINAM Institute, UMR 6213, CNRS-University Bourgogne Franche-Comté, 16, Route de Gray 25030 BESANCON Cedex France

Abstract. In this paper, we study the influence of annealing on the performance of Resistive Temperature Detectors made from Nickel thin films. The aimed application is heat flux sensing. The substrate is made of Borofloat glass with a Chromium adhesive layer. Several annealing temperatures between 150°C and 300°C are applied to this assembly. The thin films as deposited and after annealing are analyzed through SEM images. The evolution of the resistance and the temperature coefficient of the sensor are discussed. An annealing temperature is selected that ensures the repeatability of measurements.

1 Introduction

Resistive Temperature Detectors (RTD) are widely used as sensors in numerous applications. Several metals can be used for the thin film RTDs: platinum, copper or nickel. Nickel has the advantage of being more cost-effective than platinum, therefore it is suitable for wide diffusion sensors or disposable applications as in biomedical sensing [1][2]. Some authors worked on heat flux sensors using thin film nickel RTDs [3][4][5]. In heat flux sensing, the heat flux can be derived from the difference of two temperature measurements. Thus, in this application, any variation in temperature leads to a large discrepancy in the indicated heat flux. Therefore, temperature sensing has to be accurate and reliable, so the properties of the film, resistivity and Temperature Coefficient of Resistance (TCR), have to be stable. One of the method to ensure this stability is to perform an annealing of the film. This method has been widely studied in thin film Platinum [6-10] and more scarcely for Nickel [11].

Annealing leads to different phenomena that happen at different conditions and temperatures: decrease in micro-strain, raise in crystallinity, diffusion and oxidation of the sublayer if any. Therefore, it is useful to gain knowledge on the influence of an annealing process on the resistivity and TCR of a Nickel RTD associated with a sublayer and substrate. In this paper, we focus our work on the influence of low temperature annealing on the characteristics of a Nickel RTD designed for a heat flux sensing application up to 200°C. The RTD concept, practical design and microfabrication technique are presented in the first section. In the second section, the characterization of the thin films through profilometry and Scanning Electron Microscopy (SEM) is presented. The experimental setup for the
temperature and resistance measurement is detailed in the following section. Finally, experimental results for the temperature measurement without and with annealing are presented and discussed.

2 Concept, design and microfabrication of the sensor

2.1 Concept

Gradient heat flux sensors are based on the application of Fourier’s law:

\[ q = -\lambda \frac{\Delta T}{e} \]

where the temperature difference \( \Delta T \) is measured across a substrate of known thickness \( e \) and conductivity \( \lambda \). The thermal resistance of the sensor \( e/\lambda \) is usually small in order to avoid disturbance of the thermal environment.

The resistivity \( \rho \) of thin metal film can be expressed by Matthiesen’s rule \([11]\):

\[ \rho = \rho_T + \rho_r \]

where \( \rho_T \) is the temperature-dependant term and \( \rho_r \) is the residual resistivity term. \( \rho_r \) includes the contribution from surface scattering, impurities, magnetic disorder effects, grain boundaries and intragranular effects. Thus, using thin film metal RTD in a gradient heat flux sensor requires that the variation of resistivity is only linked to the temperature variation and not to uncontrolled variations of the residual resistivity. If not, this variation divided by a small thermal resistance would lead to large heat flux errors. Assuming the resistivity evolves linearly with temperature:

\[ \rho(T) = \rho_0 \left( 1 + \alpha(T - T_0) \right) \]

where \( \rho_0 \) is the resistivity of the material at the reference temperature \( T_0 \). This implies that the resistance of the RTD can be expressed as:

\[ R(T) = R_0 \left[ 1 + \alpha(T - T_0) \right] \]

where \( R_0 \) is the resistance at the reference temperature \( T_0 \). The TCR \( \alpha \) can be expressed as:

\[ \alpha = \frac{1}{R_0} \frac{dR}{dT} = \frac{R(T) - R_0}{R_0(T - T_0)} \]

1.2 Design and fabrication

The heat flux sensor is made from two RTDs deposited on each side of a Borofloat glass substrate (Figure 1). It was manufactured in the MIMENTO cleanroom at FEMTO-ST institute. A 20 nm sublayer made of chromium ensures the adhesion between the substrate and the Nickel film. The thickness of the Nickel track is about 220 nm. The total sensor dimensions are 5 x 5 x 0.5 mm\(^3\), with a sensing area of 2 x 2 mm\(^2\). More details on the sensor can be found in previous works where platinum RTDs were used \([12]\). The properties of the materials used in the sensor are summarized in Table 1. At the end of the fabrication process, an annealing at different temperatures \( T_A \) from 150°C to 300°C during 6h was applied to the sensors (Figure 2).
2 Characterization of the thin films

The thickness of the Nickel tracks was measured with an accuracy of 0.1% by a TENCOR profilometer (DEKTAK 150). Two examples of profiles on either side of the substrate are shown in Figure 4. The deposition thickness was of the order of 0.220 μm and the track width is 60 μm which is in agreement with the design.

The characterization of the films was also achieved by SEM. The film surfaces were observed as deposited and after annealing at 200°C and 300°C. We did not find any structural differences as a function of different annealing temperatures (Figure 4).

![Figure 1 Heat flux sensor scheme (a) and photograph (b)]

| Property                                      | Borofloat Glass | Nickel  | Chromium |
|-----------------------------------------------|-----------------|---------|----------|
| Resistivity $\rho$ (Ω.m) at 25 °C             | $6.9 \times 10^{-8}$ | 13.2 $\times 10^{-8}$ |          |
| TCR $\alpha$ (C$^{-1}$) at 25-500 °C (bulk)   | 0.0068           | 0.00214 |          |

Figure 2 Annealing profile

Figure 3 Profiles of the tracks of nickel films.

Figure 4. SEM pictures of Ni/Cr films a/ as deposited – b/ annealed 200°C – c/ annealed 300°C
3 Experimental results

3.1 Temperature calibration bench

The calibration bench was composed of a portable calibration oven (550 Gemini LRI), a highly accurate Pt100 platinum reference probe (0.005 °C), a reference thermometer (PHP 601) (Figure 5a) and two precision 6½ digits multimeters Keithley 2100. The RTDs were connected with the 4-wire method with constantan wires. An applied current of 0.1 mA was applied in order to minimize self-heating. At each temperature step, 50 observations of the resistance value were made. The uncertainty analysis led to a value of 0.14 °C.

Figure 5 a/ Calibration test bench. b/Calibration curves of RTDs as deposited and with annealing

3.2 Characteristics of the sensor without and with annealing

3.2.1 Characteristics of the sensor without annealing

The initial calibration curves of the nickel RTDs R₁ and R₂ as deposited (without annealing) are presented in Figure 5b. The difference in the values of resistance R₁ and R₂ can be explained by the difference of film deposition from one side to the other one (Figure 2). The observed averaged resistivity of both resistances deduced from the track dimension was 1.31x10⁻⁷ Ω.m at 35°C and the average TCR was 0.00143 °C⁻¹ (Table 2). As expected in thin films, we observe an increase in resistivity compared to bulk nickel (Table 1) and a decrease in TCR. As analysed by Johnson [11], the main contribution of this increase is probably due to the raise in grain boundary scattering.

| Conditions               | R1     | R2     |
|-------------------------|--------|--------|
| As deposited            | 0.00149| 0.00138|
| As deposited : average value | 0.00143|
| After annealing         | 0.00156| 0.00149|
| After annealing : average value | 0.00152|

Table 2. TCR as deposited and after annealing at 250°C

In order to investigate the repeatability of the measurement, 3 successive tests on the same sensor were performed after the calibration (Figure 6 Test1 to Test3). We observed a drift
the measurement up to 4.1°C. Thus, to overcome this problem, different annealing temperatures $T_A$ from 150°C to 300°C as described in § 1 were applied to the sensors.

![Figure 6](image)

**Figure 6** Absolute errors of the RTDs ($R_1 \ a/ - R_2 \ b/$) as deposited and with annealing at 250°C

### 3.2.2 Characteristics of the sensor with annealing

The evolution of the resistance at room temperature of the sensor versus the annealing temperature is plotted in Figure 7. We observed a decrease down to 200°C, then an increase. This phenomenon was also reported by Schmid and Seidel [10] and Zribi et al. [12] on a Pt/Ti and Pt/Cr film respectively. On the other hand, Johnson only reported a decrease in resistivity on a Ni film without sublayer. Concerning the decrease in resistance observed after annealing at the lowest temperatures, we make the hypothesis that the main contribution is linked to the release of the micro stress induced by the deposition process, as observed by Sreemani and Sen for platinum [9]. A raise in crystallinity due to the annealing can also lead to a decrease in resistivity. Concerning the increase observed above 200°C, we make the hypothesis that the increase is due to the diffusion and oxidation of the chromium buffer layer along grain boundaries. This phenomenon was frequently observed for Pt thin films deposited on a sublayer (see for example [8]).

![Figure 7](image)

**Figure 7** Influence of the annealing temperature on the resistance.

The annealing temperature of 250°C was thus chosen in order to benefit from the release of stress in the film. Then, the calibration was performed using the same experimental conditions. We observed a decrease in resistance, thus in resistivity. We also observed a 6% increase in the average TCR up to 0.00152 °C$^{-1}$. The absolute errors versus the reference temperature are displayed in Figure 6. We observed that the annealing also led to a high improvement in the repeatability of the results. The absolute errors are reduced to a maximal
error of 0.44°C instead of 4.12°C without annealing. Therefore, this choice of annealing temperature led to an improvement of the RTDs.

Conclusion

In this paper, we have presented the influence of low temperature annealing on nickel RTDs deposited on a glass substrate with a chromium sublayer. The aimed application is heat flux sensing where repeatable and accurate measurement are needed. We applied different annealing temperatures from 150°C to 300°C for 6 hours and showed that the resistivity decreased down to 200°C then increased. The films were analysed though SEM images and no major changes were observed between as deposited and annealed films. We made the hypothesis that the decrease was due to the reduction of stress in the film, and the increase was due to the oxidation and diffusion of the chromium sublayer. Applying a 250°C annealing led to repeatable temperature measurements and an increased TCR. The achieved performance seems compliant with the aimed application oh heat flux sensing.

This work has been supported by the EIPHI Graduate School (contract "ANR-17-EURE-0002") and was partly supported by the French RENATECH network and its FEMTO-ST technological facility.

[1] E.J.P. Santos and I.B. Vasconcelos, in 2008 26th International Conference on Microelectronics, (2008).
[2] B. Davaji and C. H. Lee, Biosens Bioelectron, 59,120 ( 2014).
[3] M. Hayashi, A. Sakurai, and S. Aso, Memoirs of the Faculty of Engineering, Kyushu University, 44, 113 (1984).
[4] A.H. Epstein, G.R. Guenette, R.J.G. Norton, and Y. Cao, Rev. Sci. Instrum., 57, no 4, 639 (1986).
[5] H. Mocikat and H. Herwig, Sensors, 9, no.4, 3011(2009).
[6] D. Resnik, J. Kovac, D. Vrtacnik, M. Godec, B. Pecar, and M. Mozek, Thin Solid Films, 639, 64 (2017).
[7] V. Guarnieri, L. Biazi, R. Marchiori, and A. Lago, Biomatter, 4, no. 1, e28822 (2014).
[8] R. Tiggelaar, R. Sanders, A. Groenland, and J. Gardeniers, Sensor Actuat. A: Phys., 152, no.1, (2009).
[9] M. Sreemany and S. Sen, Appl. Surf. Sci., 253, no. 5, 2739 (2006).
[10] U. Schmied and H. Seidel, Thin Solid Films, 516, no. 6, 898 (2008).
[11] B. Johnson, J. Appl. Phys., 67, no. 6, 3018 (1990).
[12] A. Zribi, M. Barthès, S. Bégot, F. Lanzetta, J. Rauch, and V. Moutardier, Sensor Actuat. A: Phys., 245, 26 (2016).