Pressure-Induced Crystallization and Phase Transformation of Para-xylene

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Static pressure is an alternative method to chemical pressure for tuning the crystal structure, bonds, and physical properties of materials, and is a significant technique for the synthesis of novel materials and fundamental research. In this letter, we report the crystallization and phase transformation of p-xylene under high pressure. Our optical micrographic observations and the appearance of lattice modes in the Raman and infrared (IR) spectra indicated that p-xylene crystallizes at ~0.1 GPa. The X-ray diffraction (XRD) pattern at 0.84 GPa suggests that the crystallized p-xylene had a monoclinic phase with the Cc(9) space group. The sharp shrinkage of the lattice at ~13 GPa and the solid state of the decompressed sample we observed suggests a new crystalline phase of p-xylene. The in situ XRD showed that the new crystalline phase was still a monoclinic structure but with a different space group of C2(5), indicating that a phase transition occurred during further compression. The mass spectrometry experiment confirmed phase transition polymerization, with mainly trimer and tetramer polymers. Our findings suggest an easy and efficient method for crystallizing and polymerizing p-xylene under high pressure.

High-pressure physics and chemistry offer innovative strategies to synthesize new materials for various applications1–3. One such strategy is the crystallization of liquid as an alternative to crystallization at low temperature. The pressure-induced synthesis of crystals is industrially important because eliminating catalysts produces more pure products. Thus, pressure is a powerful tool to produce crystalline organic compounds. Many examples of this have been investigated by several groups4–6. Piermarini et al. found that carbon tetrachloride solidifies into solid Ib at 0.13 GPa, transforms into another solid phase II at 0.4 GPa, and finally Phase III at 0.7 GPa7. Intensive research was conducted on the carbon tetrachloride using XRD under high pressure to define its detailed structures.

Para-xylene (p-xylene) is an aromatic hydrocarbon consisting of a benzene ring with two methyl substituents at opposite positions. The symmetry is D2h (this designation ignores the methyl group hydrogen atoms). Polymerized p-xylene is an industrially important polymer, which is primarily used as a basic raw material in the manufacturing of terephthalic acid (TPA), purified terephthalic acid (PTA), and dimethyl-terephthalate (DMT). These, in turn, are used to manufacture polyethylene terephthalate (PET)-saturated polyester polymers. These polymers are lightweight, shatter-resistant, and have high tensile strength. With the increasing global demand for TPA, the need for polymerized p-xylene production is also growing rapidly. The high yield and high stability of the polymer make this process appealing for large-scale applications. However, the crystallization, polymerization, and characterization of p-xylene under high pressure have not yet been reported.

Here, we report our crystal growth observations and pressure-induced phase transformation of p-xylene up to ~31 GPa in a diamond anvil cell (DAC) at room temperature (24–28 °C in our lab). As the pressure was carefully manipulated, p-xylene crystallized at ~0.1 GPa. We also investigated the crystallized p-xylene using XRD. Our

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preliminary Raman and IR analysis indicated that the pressure-induced phase transformation (polymerization) occurred around 13 GPa. Remarkably, the polymerized solid phase was quenchable to ambient conditions and confirmed by mass spectrometry experiments on the decompressed samples.

Results and Discussions
Liquids usually crystallize at a defined pressure. Many organic solvents have been crystallized via pressure, including odd-numbered normal alkanes (carbon number: 9, 11 and 13), cyclohexane and carbon tetrachloride. However, the properties of crystallized \( p \)-xylene remain unknown. Here, we studied the pressure-induced crystallization of \( p \)-xylene with optical microscopy and XRD at room temperature. The \( p \)-xylene was compressed above the freezing pressure of the liquid. Then, we obtained a single crystal through controlling the pressure by repeating the decompression and compression cycles. The freezing pressure of \( p \)-xylene was \( \sim 0.1 \) GPa. Images of the equilibrium forms of the \( p \)-xylene grown in the DAC are shown in Fig. 1(a–c). Finally, the crystals fully accommodated the DAC at a higher pressure (0.3 GPa), as shown in Fig. 1(d).

In order to better characterize the crystallized \( p \)-xylene, XRD patterns were performed under high pressure and room temperature. The first trace was measured at 0.84 GPa. Independent Raman measurements of \( p \)-xylene from \( \sim 0.1 \) to \( \sim 1.1 \) GPa were performed, and no detectable variation was observed between 0.1 and 0.84 GPa. Therefore, the sample at 0.84 GPa was the same as the crystal formed at \( \sim 0.1 \) GPa.

The XRD patterns and typical two-dimensional (2D) diffraction image of the \( p \)-xylene obtained at 0.84 GPa are shown in Fig. 2. The appearance of diffraction rings in the resulting 2D diffraction image suggested a crystal phase and confirmed that the \( p \)-xylene crystallized completely at this pressure. The diffraction peaks were analyzed and the \( hkl \) reflection lines are highlighted in Fig. 2. The results indicated that the crystallized \( p \)-xylene had a monoclinic symmetry with cell dimensions of \( a = 10.864 \) Å, \( b = 9.138 \) Å, \( c = 13.204 \) Å, and \( \gamma = 137.54^\circ \). This phase differed from that determined at low-temperature in another study; possibly due to the different pressure and temperature conditions of the two studies or the perdeuteration (D) influence.

XRD experiments of \( p \)-xylene were performed up to 30.7 GPa at room temperature (Fig. 3). All of the diffraction peaks broadened as the pressure increased, possibly because of the non-hydrostatic pressure in the sample chamber. The Bragg peaks shifted towards high angles (low \( d \)-spacing) due to lattice compression. During...
compression up to 10.0 GPa, no obvious change besides the peak broadening was observed in all diffraction peaks, indicating that the sample remained in a low-pressure monoclinic phase. It is worth noting that there were clear changes in the intensity of the low-angle Bragg peaks (compare $-112$ with $020$ and $002$, and $111$ with $-113$).

The diffraction peaks' relative intensities change was most probably due to the preferred orientation of the crystals at high pressure.

As the pressure reached 13.5 GPa, the intensities of $(111)$ and $(-222)$ started to decrease and eventually disappeared at 17.2 GPa, suggesting the sample completely transformed into a new phase. In the range of 13.5–17.2 GPa, two phases coexisted. Our analysis of the XRD results proposes that the sample was still in a monoclinic structure but with a different space group of $C_2(5)$ at 17.2 GPa, and the cell dimensions were $a = 4.487$ Å, $b = 9.303$ Å, $c = 4.281$ Å, and $\gamma = 138.91^\circ$. All the diffraction peaks of this new monoclinic phase (at 17.2 GPa) are also indexed in Fig. 3. The basic regular molecular arrangement in $p$-xylene forms a lattice, with the $p$-xylene molecules occupying the lattice site. The parallel arrangement of the hexagonal faces of the benzene rings with a slight shift should be just one of the arrangement forms in $p$-xylene. However, the limited information obtained from powder XRD did not allow us to determine a reliable crystal structure (symmetry, atomic position etc.). As the pressure increased above 19 GPa, the intensity of the XRD pattern became weaker and weaker, because the sample thinned at high pressure with pressure-induced amorphization.

Remarkably, the XRD data of the decompressed sample showed obvious diffraction peaks, indicating that the recovered sample was crystalline, unlike its initial liquid state. The state of organic compounds is closely related to the number of C atoms they contain, for example, alkane will not be solid under ambient conditions if the molecule contains less than 16 C atoms. This suggests that the decompressed sample could be polymerized $p$-xylene since monomeric $p$-xylene only contains 8 atoms. Therefore, polymerization is likely to be the driving force of the phase transition around 13.5 GPa.

Figure 3. Pressure evolutions of the X-ray diffraction patterns of $p$-xylene up to 30.7 GPa. The Bragg peaks' indexes and the cell parameters at 0.8 and 17.2 GPa are shown in the figure, respectively.
The pressure dependence of the \( d \)-spacings from 0.8–30.7 GPa is summarized in Fig. 4. It is clear that several \( d \)-spacings shrank sharply at around 13.5 GPa, which is consistent with the phase transition pressure, giving an insight into the transformation at this pressure. The molecules moved closer and closer as the pressure increased. The neighboring molecules started to transform as the pressure reached a critical value, and caused a much sharper decrease for some of the \( d \)-spacings, which may be due to polymerization in certain lattice directions.

Raman spectroscopy is a useful tool for examining the structures of materials\(^{12-17}\). While there have been studies of \( p \)-xylene with Raman spectroscopy under ambient conditions\(^{18,19}\), investigations at high pressure have not yet been reported. Here, we performed Raman spectroscopy of \( p \)-xylene (80–3200 cm\(^{-1}\)) up to 30.9 GPa at room temperature. Figure 5 shows the Raman spectra and major vibrational modes of \( p \)-xylene under ambient conditions. The Raman peaks for \( p \)-xylene are listed in Table 1\(^{20}\) and the Raman spectra are grouped based on these assignments, as shown in Fig. 5. Group I is assigned to the ring deformation modes, and three peaks were observed in this region. The skeletal vibration centered at 644 cm\(^{-1}\) is related to Group II. In substituted benzene ring compounds, the C–H out-of-plane bending vibrations\(^{21} \) give rise to bands in the region of 700–1000 cm\(^{-1}\). Four bands (Group III) were detected in this region and assigned to the C–H out-of-plane bending of the benzene ring. Group IV is assigned to the symmetry and anti-symmetry deformation of CH\(_3\), and Group V are those modes stretching the C=C of the benzene ring. The bands in the region of 2800–3200 cm\(^{-1}\) are group VI and assigned to the CH\(_3\) and C–H stretching vibrational modes.

Figure 6(a–c) presents the Raman spectra of \( p \)-xylene at various pressures. To better describe the pressure shift of the observed vibrational modes, the pressure dependence of the bands is summarized in Fig. 7(a–c). At 0.9 GPa, a new band appeared at 129 cm\(^{-1}\) (\( \nu_4 \)) and was assigned to the lattice mode. This confirms the crystallization of...
p-xylene at high pressure. These changes are consistent with the crystallization observed by microscopy [see Fig. 1].

When the Raman spectra exceeded 13.8 GPa, the intensity of the C−H, CH₃, and C=CH stretching modes decreased, and the full width at half maximum (FWHM) increased. A band splitting appeared at 1657 cm⁻¹, which was assigned to the anti-symmetry stretching of the benzene ring C=C. The enlarged view of the band splitting in the region of 1550–1750 cm⁻¹ for the selected pressures is presented in Fig. 6(c). The polymerized phase was obtained from a number of triple- and double-bonded hydrocarbon compounds, such as acetylene and ethylene, at relatively low pressures (a few GPa). When the C=C bonds are involved in a ring, the system is more stable and expected to polymerize above 10 GPa. Therefore, the Raman changes above 13.8 GPa could be related to a new crystalline polymerized phase, as verified by the mass spectrometry investigation on the decompressed samples. These Raman changes were also consistent with the lattice shrinkage observed in the XRD. Furthermore, the ν₁ band disappeared, bands ν₂, ν₄, and ν₅ became very weak and broad [see Fig. 6(a)]. The peak weakening and broadening in the spectra suggested the presence of p-xylene strain, confirming the pressure-induced phase transition identified by XRD. The decompressed spectrum is clearly different from that at 1 atm as shown in Fig. 8, confirming that the transition is irreversible.

Infrared spectroscopy is also a powerful tool for characterizing polymers²²,²³. We performed in situ high-pressure IR to investigate the pressure-induced crystallization and transformation processes, as shown in Fig. 9. Based on the assignments of the IR spectra in Table 1²⁴, they were divided into three regions: (i) The skeletal and CH₃ stretching vibration; (ii) The CH₃ deformation and C=C stretching vibration; and (iii) The CH₃ and C−H stretching vibration. At 0.9 GPa, four new peaks were observed, marked by asterisks. The presence of new peaks could arise from the enhancement of the intermolecular interactions due to crystallization²⁵, which was confirmed by a micrographic observation and XRD²⁶. The peak intensity increases marked by rhombuses in Fig. 9(c) were also due to the crystallization of p-xylene²⁷. As pressure increased, obvious intensity changes were found, especially for the CH₃ and C−H stretching vibration, as shown in Fig. 9(c). At 13.0 GPa, the CH₃ stretching, CH₃ deformation, and C=C stretching bands broadened, and a peak splitting appeared (marked by arrows) at 1555 cm⁻¹. This splitting became much clearer at 16.7 GPa. To our knowledge, the splitting of the vibrational mode and the softening concomitant mode of the intramolecular modes are usually attributed to the formation

| Mode  | Raman, cm⁻¹ | Intensity | IR, cm⁻¹ | Intensity | Assignments and approximate description |
|-------|-------------|-----------|----------|-----------|------------------------------------------|
| ν₄    | 312         | w         |          |           | Ring deformation                         |
| ν₅    | 386         | vw        |          |           | Ring deformation                         |
| ν₆    | 458         | s         |          |           | Ring deformation                         |
| ν₁₀   | 644         | m         |          |           | Skeletal                                 |
|       | 795         | vs        |          |           | Skeletal                                 |
|       | 1024        | w         |          |           | CH₃ stretching                           |
|       | 1042        | w         |          |           | CH₃ stretching                           |
|       | 1104        | w         |          |           | CH₃ stretching                           |
|       | 1120        | w         |          |           | CH₃ stretching                           |
| ν₁₂   | 699         | vw        |          |           | C−H out-of-plane bending                 |
| ν₁₃   | 810         | m         |          |           | C−H out-of-plane bending                 |
| ν₁₄   | 827         | vs        |          |           | C−H out-of-plane bending                 |
| ν₁₆   | 930         | vw        |          |           | C−H out-of-plane bending                 |
| ν₁₈   | 1204        | vs        |          |           | Ring rocking                            |
| ν₁₉   | 1312        | w         |          |           | CH₃ sym. deformation                     |
| ν₂₀   | 1377        | m         | 1378     | w         | CH₃ sym. deformation                     |
| ν₂₁   | 1449        | w         | 1454     | m         | CH₃ sym. deformation                     |
| ν₂₂   | 1577        | vw        | 1516     | s         | C=C sym. stretching                      |
| ν₂₃   | 1617        | m         | 1631     | vs        | C=C asym. stretching                     |
| ν₂₅   | 2559        | vw        |          |           | Combination/overtone                     |
| ν₂₆   | 2732        | w         |          |           | Combination/overtone                     |
| ν₂₇   | 2863        | w         | 2869     | s         | CH₃ sym. stretching                      |
| ν₂₈   | 2919        | s         |          |           | CH₃ asym. stretching                     |
|       | 2932        | vs        |          |           | CH₃ sym. stretching                      |
|       | 2946        | w         |          |           | CH₃ asym. stretching                     |
|       | 2975        |           |          |           | CH₃ asym. stretching                     |
| ν₃₀   | 3011        | m         | 3000     | s         | Benzene C−H stretching                   |
| ν₃₁   | 3053        | s         | 3045     | m         | Benzene C−H stretching                   |

Table 1. The Raman and infrared spectrum and assignments for p-xylene under ambient conditions. vs: very strong; s: strong; m: medium; w: weak; vw: very weak.
of new intermolecular bonds. Remarkably, the C=C stretching band at 1670 cm\(^{-1}\) disappeared and three new apparent peaks (marked by crosses) were observed at around 3210 cm\(^{-1}\), which was also due to new intermolecular bonds forming. Furthermore, we collected the IR spectra of the sample recovered from 33 GPa and compared it with the spectra at 1 atm. Note that the spectra at 1 atm and the spectra recovered from 33 GPa are different. Thus, the IR results confirm the pressure-induced transformation. The process of transformation involves the formation of new intermolecular bands, which relate to the C=C and C–H vibration of the benzene ring, and CH\(_3\) (deformation and stretching) vibration changes. This result is consistent with our Raman and XRD findings.
Next, we performed several additional independent experiments to confirm the new pressure-induced crystalline phase transformation. In these experiments, three recovered samples were obtained by recovering p-xylene from 12.6, 15.9, and 20.1 GPa at room temperature, respectively. We then compared the samples to each other. We found that the sample released from 12.6 GPa was liquid and finally evaporated when decompressed. However, the products recovered from 15.9 and 20.1 GPa were a transparent solid and crushed when the sample was poked with a needle, as shown in Fig. 10. The new phase could be sustained for a few months without further changes.

We also performed high-performance liquid chromatography (HPLC) mass spectrometry to investigate the decompressed p-xylene. The experiments were performed with a positive ion and the samples were decompressed.
from 25 GPa to ambient conditions. From the mass spectrum, a number of mass-to-charge (m/z, m is the mass and z is charge) appeared at m/z 289, 336, 390, 338, as shown in Fig. 11. Through analysis, we found that the m/z value of an oligomer is not exactly the integral multiple of the mass number of the C8H10 monomer. The random copolymerization and isomeric adjustment make the m/z of the polymer different. However, the oligomers mainly involved trimer and tetramer polymers. Therefore, we conclude that the application of pressure up to around 13 GPa does indeed lead to a new crystalline polymerized phase transformation of p-xylene.

**Conclusions**

In conclusion, the crystallization and properties of p-xylene under high pressure were investigated by optical microscopy, XRD, Raman, and IR spectroscopy. P-xylene crystallized at ~0.1 GPa with a monoclinic structure. Characteristic crystalline bands were identified by analyzing the Raman and IR spectra at 0.9 GPa. Our preliminary high-pressure Raman and IR analysis indicated that the pressure-induced polymerization occurred around 13 GPa. The new crystalline polymerized phase structure of p-xylene was characterized using results from XRD.
investigations at 17.2 GPa. Remarkably, the polymerized solid phase was quenchable to ambient conditions and confirmed by mass spectrometry experiments on the decompressed samples.

Methods

High-pressure experiments were performed with a DAC equipped with a 300 μm culet. The gaskets were pre-indented to 40–50 μm with a 100 μm hole to serve as the sample chamber. The p-xylene sample (99.999% purity from Alfa Aesar Puratronic) was loaded into the DAC at room temperature (24–28 °C in our lab). A ruby ball was loaded with the sample to perform in situ pressure measurements via the R1 ruby fluorescence band shift. The crystal growth and the new phase of p-xylene were studied with optical microscopy. Raman and IR absorption measurements were carried out using a 532 nm excitation laser and a Bruker Vertex80 FTIR spectrometer, respectively. The XRD experiments under high pressure and room temperature were performed at the BL15U1 beamline at the Shanghai Synchrotron Radiation Facility (SSRF) with a wavelength of 0.6199 Å and a CCD detector. The diffraction patterns were analyzed and integrated using the FIT2D program to obtain the 1D intensity distribution as a function of the 2θ scattering angle. The mass spectrometry (MS) experiment was performed by high-performance liquid chromatography (HPLC) using an atmospheric pressure chemical ionization (APCI) source.

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Acknowledgements

Portions of this work were performed at the BL15U1 beamline, Shanghai Synchrotron Radiation Source in China. The authors would like to thank the Shanghai Synchrotron Radiation Source for use of their synchrotron radiation facilities. This work was partially supported by the Natural Science Foundation of China (Grant No.
The work was also supported by the Program for New Century Excellent Talents in University (NCET-10-0444), Ph.D. Programs Foundation of the Ministry of Education of China (20120073110057), and the State Key Laboratory of Explosion Science and Technology, Beijing Institute of Technology (KFJJ15).

Author Contributions
L.W. and D.Q.W. designed the project; Y.Z.B., Z.H.Y., N.N.L., S.Y., K.Y., R.L., B.B.L. and L.W. performed the experiments and data analysis. L.W., Y.Z.B., and Z.H.Y. wrote the manuscript. All authors reviewed the manuscript.

Additional Information

Competing Interests: The authors declare that they have no competing interests.

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