Assemblies of gold icosahedra

G. Bilalbegović

Department of Physics, University of Rijeka, Omladinska 14, 51000 Rijeka, Croatia

Abstract

Low-dimensional free-standing aggregates of bare gold clusters are studied by the molecular dynamics simulation. Icosahedra of 55 and 147 atoms are equilibrated at \( T = 300 \) K. Then, their one- and two-dimensional assemblies are investigated. It is found that icosahedra do not coalesce into large drops, but stable amorphous nanostructures are formed: nanowires for one-dimensional and nanofilms for two-dimensional assemblies. The high-temperature stability of these nanostructures is also investigated.

Key words: Molecular Dynamics Simulation, Nanoparticles, Nanowires, Nanofilms, Self-assembly, Gold

PACS: 61.46.+w, 81.16.Dn, 81.07.-b, 36.40.-c, 61.43.Bn

Email address: goranka.bilalbegovic@zg.htnet.hr (G. Bilalbegović).

Fax: +1-775-414-7947; Tel:+385-98-480-921; Postal address (home): Grada Vukovara 43A, 10000 Zagreb, Croatia

Preprint submitted to Elsevier Science 5 February 2020
1 Introduction

Gold is very interesting for applications in nanotechnology [1,2,3,4,5,6,7,8,9,10,11]. All bare low-index surfaces of gold exhibit reconstructions and act as very good substrates for inorganic, organic and biological thin films. Nanoparticles of gold show interesting optical and electronic properties. Gold clusters arranged in one and two dimensions are important for applications, most notably as quantum-dot arrays.

Isolated gold clusters are extensively studied by experimental and theoretical methods [12,13,14]. Various amorphous, crystalline and quasicrystalline structural forms were found. Ordered high symmetry structures are Mackay icosahedral and fcc cuboctahedral configurations. A producing of cluster-assembled materials is nowadays very important. The discovery of \( C_{60} \), other fullerenes, and fullerite indicates the route towards synthesis of new materials consisting of stable clusters. Gold icosahedra are stable and therefore, they can be used as a building element of assembled materials.

To stabilize aggregates of gold clusters various methods are used, for example, the Langmuir-Blodgett technique, addition of functionalized polymers, or DNA [1,2,3,4]. Gold nanoparticles and nanorods are also successfully synthesized by the electrochemical method [5]. Recently, gold nanoparticle films were prepared by the surface vertical deposition method on a wettable substrate [6]. It is especially difficult to make and investigate short one-dimensional arrangements of gold nanoparticles. Such structures were obtained for \( Au_{55}(PPh_3)_{12}Cl_6 \) clusters [1]. These 55 atoms gold clusters surrounded by ligand shells were positioned along bundles of polymers. Networks of polymer chains decorated by
gold clusters were deposited on the substrates using the Langmuir-Blodgett technique [2]. It is possible to prepare these networks with different densities. Nanostructured Au cellular networks were produced by spin coating a solution of thiol-passivated gold nanoparticles on the silicon substrate [7]. It was found that annealing at (500 − 600) K removes thiol surfactants, but the cellular structure of bare gold nanoparticles was preserved. One-dimensional organizations of gold clusters were also made by decoration of various templates, for example carbon nanotubes [8,9,10]. Gold nanoboxes with a truncated cubic shape were obtained using silver nanocubes as templates [11]. These bare gold nanoboxes self-assembled into a close-packed two-dimensional array.

In this work the Molecular Dynamics (MD) simulation results for ordered low-dimensional aggregation of small gold icosahedra are presented. It is shown that, although a coalescence of atoms occurs at the touching point of neighboring clusters, stable one- and two-dimensional aggregates are formed at room temperatures. Assembled nanostructures are stable up to at least 800 K. These results are important for understanding of processes and development of new techniques in preparation of gold nanowires and thin films along the route described in Refs. [1,2,3,4,5,6,7,8,9,10,11]. In the following a computational method is described in Sec. II. The results of simulation and a discussion are presented in Sec. III. A summary and conclusions are given in Sec. IV.

2 Computational Method

I have used a computer simulation to elucidate the stability of low-dimensional arrangements of gold icosahedra. The classical MD method is carried out [17]. This method is very suitable for modeling of large systems on long time scales.
It is not yet feasible to study these large scale problems within ab-initio methods. It is well known that the pair potentials in classical simulations do not give a correct description of metal properties [18]. Instead, many-body potentials of the embedded-atom type are successfully applied for metals. In this work the “glue” realization of embedded atom potentials is used to describe the interparticle interactions [19]. This potential is very well tested and known to give good results for many bulk, surface and nanoparticle properties [19,13,20,21,22,23,24,25]. In contrast to several other potentials used in classical simulations, this one reproduces different reconstructions on all low-index gold surfaces [19]. Previous simulations of gold clusters using the same potential confirmed several results obtained experimentally [13]. Therefore, it is expected that simulated aggregations of gold icosahedra realistically model natural nanostructures. The temperature of the systems is controlled by rescaling particle velocities. The time step used is $7.14 \times 10^{-15}$ s.

The clusters are prepared as ideal icosahedra of 55 and 147 atoms at $T = 0$ K. After a procedure of annealing and quenching, the clusters are equilibrated for $10^5$ time steps at $T = 300$ K. Several fully equilibrated clusters are then placed in contact with one another and aligned in one or two dimensions. The contact of icosahedra in one dimension is shown in Figs. 1 (a) and (c). Two, five and ten icosahedra are positioned in one dimensions. The arrangements of $5 \times 5$ clusters are made in two dimensions. Resulting structures are equilibrated for additional $10^6$ time steps, i.e., for 7.1 ns. The assembled nanostructures are also heated up to 1400 K, in steps of 100 K. They were equilibrated for $10^6$ time steps at each temperature. The experimental bulk melting point of gold is 1336 K, whereas for the potential used in this simulation the bulk melting temperature was fitted to 1357 K [19].
3 Results and discussion

An assembly of only two icosahedra evolves into an elongated, potato-shaped nanoparticle. Figure 1 presents one-dimensional assemblies of five and ten icosahedra consisting of 55 atoms. Wormlike nanowires are obtained, as shown in Fig. 1 (b) and (d). Figure 2 shows one-dimensional assemblies of 147 atoms icosahedra after $10^6$ time steps of evolution. Nanowires in Fig. 2 are less curved than one-dimensional aggregates of 55 icosahedra. The radial density plots for nanowires shown in Figs. 1 (b), (d) and Fig. 2 are presented in Fig. 3. A radial ordering does not exist. This is in contrast with radial structures of gold nanowires studied previously in MD simulations [20,21,22,23,24,25]. For example, the starting point in Refs. [20,21,22,25] was a cylindrical nanostructure carved from regular fcc lattice of bulk gold. In MD simulations these structures evolved to multi-shell nanowires similar to multi-wall carbon nanotubes. Multi-shell gold nanowires were also observed by high-resolution transmission electron microscopy [15].

In previous simulations it was found that the best multi-wall structures exist at temperatures (800 – 900) K [21]. Nanostructures produced by self-assembling of icosahedra and simulated in this work are also studied at higher temperatures. As in previous simulations [21,25] the mean-square displacements and diffusion coefficients plots show that the melting starts above 900 K. Figure 4 presents the nanostructure assembled from five 55 icosahedra at 800 K. The side view in Fig. 4 (a) shows that this structure is much more straight than at 300 K (see Fig. 1 (b)). The cross-section in Fig. 4 (b) shows multi-shells at an initial stage of development. The annealing and quenching procedure used for a gold nanowire preparation in previous simulations [20,21,22,25] produces
much better shells. The structure assembled from ten 55 atoms icosahedra (see Fig. 1 (d)) also starts to order above the room temperature. However, the kink develops at 600 K, and the structure at 900 K evolves to a spherical solid drop. The structure assembled from five 147 icosahedra (see Fig. 2 (a)) has a nanowire morphology at (300 – 800) K and evolves to a spherical solid shape at 900 K. The nanowire made from ten 147 icosahedra (shown in Fig. 2 (b)) evolves first to irregular (“potato”) morphology at 900 K, and then transforms to the spherical solid shape at 1100 K.

Figure 5 shows two-dimensional assemblies after $10^6$ time steps of evolution at 300 K. Stable thin nanofilms are formed. Their surfaces are (111) oriented with many local defects. Transition of gold (001) nanofilms to the (111) orientation induced by local thinning was observed by the electron microscope [16]. It was found there that the (111) film expands laterally and then breaks. The smallest size of nanofilms studied in Ref. [16] was $(2 \times 2)$ nm$^2$, and the largest one was $(24 \times 14)$ nm$^2$. The thickness of the (111) film just after transition from (100) was about 2 nm. Nanofilms studied here are $(4.9 \times 4.9)$ nm$^2$ and $(7.4 \times 7.4)$ nm$^2$. They are both thinner than 2 nm. The thickness of the film made of 55 atom icosahedra is $\sim 1$ nm. The film assembled from 147 atom clusters is 1.5 nm thick. For both nanofilms this morphology is stable up to 1000 K. At higher temperatures nanofilms transform to the spherical shape.

A structural analysis of nanofilms shown in Fig. 5 is carried out via the Voronoi construction [26]. This construction is a topological generalization of the Wigner-Seitz one in solid state physics. The Voronoi polyhedron for a given atom is defined as the region consisting of all points nearer to it than to any other atom. The Voronoi construction represents local atomic configurations and disorder for a given structure. Figure 6 presents the fractional concen-
tions for the number of neighbors calculated for the atoms in nanofilms shown in Fig. 5. The particles are not assumed to be neighbors beyond the cutoff of 5 Å. The mean number of the Voronoi neighbors \(< n >\) is 14.75 for the distribution shown in Fig. 6 (a), and 14.40 for Fig. 6 (b). It is known that \(< n >\) is 6 in two dimensions, and 14 in three dimensions and for regular close packing [26]. In a MD simulation Rahman found 15.67 for the randomly distributed atoms of ideal gas, 14.45 for the liquid argon, and 14.26 for the solid argon [27]. The maximum of distribution for the number of faces is 13 for Fig. 6 (a), and 15 for Fig. 6 (b). The broadness of the distributions in Fig. 6 shows the presence of topologically defective polygons where the number of faces is not 14.

It is known that bare metal clusters deposited on a substrate diffuse and often show a tendency towards coalescence and sintering. The processes of arrangements of gold icosahedra simulated here do not produce large spherical drops at 300 K, and up to 800 K. This type of coalescence is inhibited by a limited diffusion at lower temperatures and the presence of facets on initial clusters. Similar results were found in a previous MD simulation for aggregation of two solid gold nanoparticles [28]. This was explained as a result of the presence of very small facets on spherical solid clusters. The initial clusters studied here are much more regular and facetted than in Ref. [28], where spherical (135 – 3997) atom nanoparticles were carved from a block of fcc bulk gold lattice. Large solid facets on icosahedra studied here prevent strong interdiffusion and sintering into large solid drops.
4 Conclusions

Molecular dynamics simulation of one- and two-dimensional aggregations of free bare gold icosahedra has been carried out. Assembling of bare gold clusters may be used as a method for designing and fabrication of nanowires and nanofilms. It is possible to prepare gold icosahedra in solutions, or in the gas phase using supersonic beams. Similar arrangements of gold clusters as those simulated here, may be prepared on the substrates by controlled deposition, or using STM and AFM tips.

Acknowledgements

This work has been carried under the HR-MZT project 0119255 “Dynamical Properties and Spectroscopy of Surfaces and Nanostructures”.

References

[1] D. Wyrwa, N. Beyer, and G. Schmid, Nano Letters 2 (2002) 419.

[2] T. Reuter, O. Vidoni, V. Torma, G. Schmid, L. Nan, M. Gleiche, L. Chi, H. Fuchs, Nano Letters 2 (2002) 709.

[3] R. L. Whetten, M. N. Shafigullin, J. T. Khoury, T. G. Schaaff, I. Vezmar, M. Alvarez, and A. Wilkinson, Acc. Chem. Res. 32 (1999) 397.

[4] W. D. Luedtke and U. Landman, J. Phys. Chem. 100 (1996) 13323.

[5] M. B. Mohamed, Z. L. Wang, and M. A. El-Sayed, J. Phys. Chem. 103 (1999) 10255.
[6] J. J. Diao, F. S. Qiu, G. D. Chen, and M. E. Reeves, J. Phys. D: Appl. Phys. 36 (2003) L25.

[7] J. N. O’Shea, M. A. Phillips, M. D. R. Taylor, P. Moriarty, M. Brust, V. R. Dhanak, Appl. Phys. Lett. 81 (2002) 5039.

[8] T. O. Hutchinson, Y. P. Liu, C. J. Kiely, M. Brust, Adv. Mater. 13 (2001) 1800.

[9] S. Fullam, D. Cottel, H. Rensmo, D. Fitzmaurice, Adv. Mater. 12 (2000) 1430.

[10] S. Hoeppener, L. F. Chi, H. Fuchs, Nano Letters 2 (2002) 459.

[11] Y. Sun and Y. Xia, Science 298 (2002) 2176.

[12] T. P. Martin, Phys. Rep. 273 (1996) 199.

[13] F. Ercolessi F, W. Andreoni, and E. Tosatti, Phys. Rev. Lett. 66 (1991) 911.

[14] I. L. Garzon, K. Michaelian, M. R. Beltran, A. Posada-Amarillas, P. Ordejon, E. Artacho, D. Sanchez-Portal, and J. M. Soler, Phys. Rev. Lett 81 (1998) 1600.

[15] Y. Kondo and K. Takayanagi, Science 289 (2000) 606.

[16] Y. Kondo, Q. Ru, and K. Takayanagi, Phys. Rev. Lett. 82 (1999) 751.

[17] D. Frenkel and B. Smit, Understanding Molecular Simulation, Academic, London, 1996.

[18] M. S. Daw, S. M. Foiles, and M. I. Baskes, Mater. Sci. Rep. 9 (1993) 251.

[19] F. Ercolessi F, M. Parrinello, and E. Tosatti, Philos Mag A 58 (1988) 213.

[20] G. Bilalbegović, Phys Rev B 58 (1998) 15412.

[21] G. Bilalbegović, Solid State Commun. 115 (2000) 73.

[22] G. Bilalbegović, J Phys: Condens Matter 13 (2001) 11531.

[23] E. Tosatti, S. Prestipino, S. Kostlmeier, A. Dal Corso, F. D. Di Tolla, Science 291 (2001) 288.
[24] B. Wang, S. Yin, G. Wang, A. Buldum, J. Zhao, Phys. Rev. Lett. 86 (2001) 2046.

[25] G. Bilalbegović, Vacuum 71 (2003) 065501.

[26] R. Collins, in Phase Transitions and Critical Phenomena, C. Domb and M. S. Green (Eds.), Vol. 2, p. 271, Academic, London, 1972.

[27] A. Rahman, J. Chem. Phys. 45 (1966) 2585.

[28] L. J. Lewis, P. Jensen, and J. L. Barrat, Phys Rev B 56 (1997) 2248.
Fig. 1. Atomic positions at 300 K for: (a) five assembled 55 icosahedra, (b) the nanostructure shown in (a) after $10^6$ steps of time evolution, (c) ten assembled 55 icosahedra, (d) the structure shown in (c) after $10^6$ time steps of evolution.

Fig. 2. Atomic positions at 300 K for the equilibrated nanostructures assembled from: (a) five 147 icosahedra, (b) ten 147 icosahedra.

Fig. 3. The radial density plots at $T = 300$ K for the equilibrated nanowires assembled from: (a) five 55, (b) ten 55, (c) five 147, (d) ten 147 atoms icosahedra.

Fig. 4. Atomic positions for a nanowire shown in Fig. 1 (b) after the heating and equilibration procedure at 800 K: (a) side view, (b) cross-section.

Fig. 5. MD snapshots of atomic positions at 300 K for the equilibrated nanofilms assembled from: (a) 55 atom icosahedra, (b) 147 atom icosahedra.

Fig. 6. The fractional concentrations for the number of faces of the Voronoi polyhedra in nanofilms shown in Fig. 5 and assembled from: (a) 55 atom icosahedra, (b) 147 atom icosahedra. The bars show the numbers, whereas the solid line is the envelope.
Fig. 3
Fig. 6