Effect of microslit retention on the bond strength of zirconia to dental materials

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The aim of this study was to investigate the effect of microslits formed by Nd:YVO4 laser beam machining on the bond strength between two types of zirconia, yttria-partially stabilized zirconia (Y-TZP) and ceria-partially stabilized zirconia/alumina nanocomposite (Ce-TZP/A), and porcelain or two types of resin. Zirconia disks were divided into three groups: 1) non-treated (NT); 2) blasted with alumina particles (AB); 3) microslits fabricated on a zirconia surface by laser beam machining (MS). After veneering porcelain or resins on zirconia specimens, halves of the resin specimens were thermocycled up to 20,000 cycles. The shear bond strength between porcelain and both types of zirconia was not improved by the microslits. Before and after thermocycling, the bond strength between an indirect composite resin or acrylic resin and Y-TZP with microslits was the highest. It was concluded that the microslits on Y-TZP enabled micromechanical interlocking and improved the bond strength and durability of the resins.

Keywords: Bond strength, Zirconia, Microslit, Porcelain, Resin

INTRODUCTION

In the traditional methods of restorative dentistry, metal was widely used in combination with resin or ceramic. Contrary to this, zirconia has been used in recent years as monolithic anatomically contoured crown or core-materials for ceramic crown in restorative dentistry. Zirconia has advantages such as high fracture toughness, flexural toughness and better aesthetic appearance, which makes it a viable alternative to metal.

A representative example of zirconia use in combination with other materials is zirconia-supported ceramic restoration. The use of a zirconia frame for fixed partial dentures has excellent aesthetics that are better than a metal frame because zirconia is a white colored material in the normal state. However, there are some problems of fracturing and dislodgement, although such an all-ceramic must provide longevity similar to metal-ceramic composites.

It is also possible that indirect composite resins be applied to a zirconia frame in the fabrication of resin facing restorations. Indirect composite resins are easily manipulated chair-side and in the laboratory, and can be repaired in the oral cavity. The wear resistance of indirect composite resins has also been improved by the use of fillers. Considering that porcelain is a brittle material, the demand for zirconia-composite restorations may increase in the future.

Moreover, with the advance in dental computer-aided design and computer-aided manufacturing (CAD/CAM) systems, the fabrication of removable dentures that contain zirconia frameworks, bars, and clasps has become possible. Although removable partial dentures with zirconia frameworks have room for improvement for clinical use, it is anticipated that a combination of zirconia dentures and acrylic resin will be used in the future because of the biocompatibility and corrosion resistance of these materials. Therefore, it is important to establish the strength and stability of the adhesive bond between zirconia and acrylic resin.

In the bonding between zirconia and each material, improvement of the bond strength plays an important role in the component longevity. For example, some primers are effective to improve bond strength between resin and zirconia. Nevertheless, when considering a specific resin, mechanical interlocking is required for bond stability. It is expected that mechanical interlocking is one of the important methods to achieve strong adhesion. In resin-metal restoration, the use of retention beads is a popular method used to achieve this purpose by providing undercuts. However, milling systems are not able to manufacture such microstructure. Although there have been a number of reports concerning zirconia surfaces roughened by alumina blasting, acid etching, or porous structure, there is little information available on bonding zirconia with a regular arrangement microstructure to other composite materials.

Another approach is laser machining, where it is unnecessary to consider consumption of burs, and fine processing can be accurately performed. There have been a number of studies on laser irradiation of zirconia. Henriques et al. investigated the strength...
of the bond between zirconia and porcelain following structuring of the zirconia surface using neodymium-doped yttrium aluminum garnet (Nd:YAG) laser irradiation. They found that laser structuring increased the shear bond strength (SBS) by up to 75%. In addition, they reported that no phase transformation occurred as a result of laser heating. On the other hand, Noda et al.18 stated that Nd:YAG dental laser irradiation is not suitable for welding zirconia because it induces surface cracking. Machining using a neodymium-doped yttrium orthovanadate (Nd:YVO4) laser has also been reported19. It was concluded that an industrial Nd:YVO4 laser was capable of fabricating a coping directly from fully sintered yttria-partially stabilized zirconia (Y-TZP). Such lasers have a short pulse width and a power density that is several hundred times higher than that for a dental Nd:YAG laser. It is therefore considered possible that the strength of the bond between zirconia and veneering materials could be improved by fabricating microslits in the zirconia using a computerized numerical control (CNC) Nd:YVO4 laser machine. The purpose of the present study was to evaluate the effect of these microslits on the bond strength between two types of zirconia and porcelain or two types of resin.

MATERIALS AND METHODS

Specimen preparation and surface treatment

The materials used in this study are presented in Table 1. Two types of tetragonal zirconia polycrystal (TZP), Y-TZP and ceria-partially stabilized zirconia/alumina nanocomposite (Ce-TZP/A), were used. Y-TZP (Cercon Base, Degudent, Hanau-Wolfgang, Germany) and Ce-TZP/A (P-NANO ZR, Panasonic Healthcare, Tokyo, Japan) were sintered for 2 h at 1,350 and 1,450°C, respectively, to produce disks with a diameter of 10 mm and a thickness of 2.5 mm. One surface of the disks was polished with 600-grit silicon carbide abrasive paper.

Zirconia disks were divided into three groups according to the surface treatment performed: (1) non-treated (NT), (2) blasted with alumina particles (AB), and (3) with fabricated the microslits (MS). Alumina-blasting of the AB group was performed on the zirconia surface with 50 µm alumina particles (Cobra, Renfert, Hilzingen, Germany) for 10 s at a pressure of 0.3 MPa and at a distance of 10 mm between the nozzle (Basic quattro, Renfert) and the surface. The microslits on the MS group samples were fabricated as reticular slits (width and pitch: 40 µm, depth: 25µm) using a CNC laser machine (LASERTEC 40, DMG MORI, Nagoya, Japan). The laser medium was Nd:YVO4, and the laser irradiation parameters were as follows: frequency of 70 kHz, wavelength of 1,065 nm, 60 mm from the surface, and normal incidence. After laser machining, a thin surface layer became black because laser irradiation reduced oxygen content on the surface of zirconia18. Since black color is aesthetically undesirable for clinical use, the specimens were then heated at a rate of 55°C/min and held at 1,000°C for 5 min to recover oxygen17 and restore white color. Figures 1 and 2 show SEM images of surface of Y-TZP and Ce-TZP/A before veneering.

Material veneering procedure

The procedure for firing porcelain to zirconia was

Table 1 Materials used for this study

| Material               | Composition                  | Manufacturer                  | Lot No.  |
|------------------------|------------------------------|-------------------------------|----------|
| Zirconia               |                              |                               |          |
| Cercon Base (Y-TZP)    | ZrO2, Y2O3, HfO2, Al2O3, SiO2| Degudent, Hanau-Wolfgang, Germany | 18007813 |
| P-NANO ZR (Ce-TZP/A)   | ZrO2, Al2O3, CeO2, HfO2     | Panasonic Healthcare, Tokyo, Japan | G2222    |
| Porcelain              |                              |                               |          |
| Cercon Ceram Kiss      | Selenium, Feldspathic porcelain | Degudent                   | 91685    |
| Paste Liner            |                              |                               |          |
| Dentin                 | Feldspathic porcelain        |                               | 93629    |
| Indirect composite resin|                             |                               |          |
| Gradia                 |                              | GC, Tokyo, Japan             |          |
| Foundation Opaque      | UDMA, SiO2                   |                               | 130902B  |
| Opaque                 | UDMA, SiO2                   |                               | 1310152  |
| Dentin                 | Filler, UDMA, SiO2, grass powder |                               | 1307171  |
| Acrylic resin          |                              |                               |          |
| PalaXpress ultra       |                              | Kulzer, Hanau, Germany       |          |
| Powder                 | PMMA, etc.                   |                               | 12030    |
| Liquid                 | MMA, etc.                    |                               | 10223    |
| Primer                 |                              |                               |          |
| Alloy primer           | VTD, MDP, Acetone            | Kuraray Noritake Dental, Tokyo, Japan | 190025  |

UDMA=urethane dimethacrylate; PMMA=Polymethyl methacrylate; MMA=methyl methacrylate; VTD=6-(4-vinylbenzyl-n-propyl)amino-1,3,5-triazine-2,4-dithiol, or -2,4-dithione tautomer; MDP=10-methacryloyloxydecyl dihydrogen phosphate; PETP=pentaerythritol tetrakis(3-mercaptopropionate)
Fig. 1  SEM images of surface of Y-TZP before veneering (100×).

Fig. 2  SEM images of surface of Ce-TZP/A before veneering (100×).

Table 2  Firing schedule of porcelain

|              | Pre-heating (°C) | Drying time pre-heating (min) | Heating rate (°C/min) | End temp. (°C) | Holding time (min) | Vacuum (hPa) |
|--------------|------------------|-------------------------------|-----------------------|----------------|-------------------|--------------|
| Paste liner 1| 575              | 8:00                          | 55                    | 970            | 1:00              | 50           |
| Paste liner 2| 575              | 8:00                          | 55                    | 960            | 1:00              | 50           |
| Dentin       | 450              | 5:00                          | 55                    | 830            | 1:30              | 50           |

conducted as follows. After a layer of liner material (Cercon Ceram Kiss, Degudent) was applied to the area defined by tape with a circular 5 mm diameter hole, the tape was removed from the surface of the specimen. The specimen was then transferred to a firing tray and fused in a porcelain furnace, according to the manufacturer’s instructions given in Table 2. After firing the liner, dentin porcelain (Cercon Ceram Kiss, Degudent) was applied onto the liner surface using a cylindrical vinyl mold (6 mm internal diameter, 2.5 mm long, and 1 mm wall thickness). The mold was removed after drying. Dentin porcelains were fired in a porcelain furnace (Cerafusion VPF, J Morita, Osaka, Japan) according to the manufacturer’s instructions given in Table 2.

The procedure for bonding indirect composite resin to zirconia was conducted as follows. After application of the primer to the zirconia surface in the same way as that for the indirect composite resin, denture base resin (PalaXpress ultra, Kulzer) was mixed at a powder/liquid ratio of 10 g/7 mL and poured into a brass mold (6 mm internal diameter, 2 mm long, and 1 mm wall thickness), then fixed by double-sided tape with a circular 5 mm diameter hole. The specimens were then polymerized in a pressure vessel (Palamat practic ELT, Kulzer) according to the manufacturer’s instructions (Polymerization time: 30 min, water temperature: 55°C, pressure: 2 bar).

All specimens were immersed in distilled water at 37°C for 24 h. This state was defined as 0 thermocycles. Indirect composite resin and acrylic resin specimens were placed in a thermocycling apparatus (Thermal cycler, Nissin Seiki, Hiroshima, Japan) and cycled between 4 and 60°C in water with a 1 min dwell time per water bath for 20,000 cycles.

Shear test and statistical analysis
Specimens veneered with porcelain, indirect composite resin, and acrylic resin (n=10) were seated in a shear testing jig during testing. Loads for shear tests were applied using a universal testing machine (AGS-5kNJ, Shimadzu, Kyoto, Japan) at a crosshead speed of 0.5
mm/min. SBS was calculated by dividing the peak load (N) by the bonded area. The values for each group were compared using the two-way analysis of variance (ANOVA) at a significance level of 0.05. After ANOVA, multiple comparisons were performed using the Scheffe’s test. The statistical significance level was set at 0.05. In addition, to analyze the influence of the thermocycling, the results for 0 and 20,000 cycles of an identical surface treatment were compared using Mann-Whitney’s U test with the value of statistical significance set at 0.05 for each treatment. All analyses were performed using SPSS 15.0 for Windows (SPSS Japan, Tokyo, Japan).

Failure analysis
After the shear bond tests, failure modes were categorized as adhesive failure at the zirconia-veneer material interface (A), cohesive failure at the inside of veneer material (C), or a mixture of adhesive and cohesive failure (AC). The bonding surfaces of specimens used for the shear bond test were observed using scanning electron microscopy (SE-8000, Keyence, Osaka, Japan) at a magnification of 500×.

RESULTS
Table 3 shows the SBSs of porcelain veneered to zirconia. Among the Y-TZP specimens, the SBS ranged from a maximum of 26.2 MPa in the MS group to a minimum of 23.4 MPa in the NT group. However, there was no significant difference between groups. On the other hand, the SBS for Ce-TZP/A ranged from a maximum of 24.5 MPa in the MS group to a minimum of 20.1 MPa in the AB group. However, no statistical significant differences were found. Failure modes after shear tests are summarized in Table 4. The MS groups of both types of zirconia showed cohesive failure (C), and SEM observations of the specimens used for the shear test revealed that residual porcelain penetrates into the microslits of both MS groups (Figs. 3 and 4).

Table 5 shows the SBSs of indirect composite resin veneered to zirconia. For the mean SBS values of Y-TZP before and after thermocycling, the MS group was highest, and the AB group was significantly higher than the NT group. On the other hand, for the mean SBS values of Ce-TZP/A, there was a significant difference among all groups and the AB group had the highest mean SBS followed by the MS and NT groups. Failure modes after shear tests are summarized in Table 6. No cohesive failure (C) was found in the NT and AB groups among both types of zirconia. For Y-TZP, the MS groups exhibited cohesive failure (C) both before and after the thermocycles. However, the MS group of Ce-TZP/A tended to exhibit mixed failure (AC) before the thermocycles and adhesive failure (A) after the thermocycles. SEM observations revealed that the microslits on Y-TZP were filled with resin before and after thermocycles; however, the resin in the other groups was either completely or partially debonded (Figs. 5 and 6).

Table 7 shows the SBSs of acrylic resin veneered to
Table 5  Shear bond strengths of indirect composite resin veneered to zirconia

| Groups  | 0 Thermocycles | 20,000 Thermocycles | S  | Reduction |
|---------|----------------|----------------------|----|-----------|
|         | Mean | SD  | Category | Mean | SD  | Category | S  | Reduction |
| Y-TZP   |      |      |          |      |      |          |    |           |
| NT      | 21.7 | 2.4 | a        | 0    | 0   |          | S  | 100%       |
| AB      | 27.2 | 4.8 | b        | 27.9 | 4.4 | d        |    | −2.6%      |
| MS      | 32.5 | 5.2 | c        | 35.9 | 5.9 | e        |    | −10.5%     |
| Ce-TZP/A|      |      |          |      |      |          |    |           |
| NT      | 15.3 | 8.0 | A        | 0    | 0   |          | S  | 100%       |
| AB      | 30.0 | 3.7 | C        | 24.6 | 3.7 | D        |    | 18.0%      |
| MS      | 21.9 | 4.3 | B        | 13.6 | 6.5 | E        |    | 37.9%      |

SD: Standard deviation; Category: Identical letters indicate that values are not statistically different (p>0.05); S: Significant difference between pre- and post-thermocycling bond strengths (p<0.05).

Table 6  Failure modes of indirect composite resin after shear bond testing

| Groups   | 0 Thermocycles | 20,000 Thermocycles |
|----------|----------------|---------------------|
|          | A  | AC | C  | A  | AC | C  |
| Y-TZP    |    |    |    |    |    |    |
| NT       | 0  | 10 | 0  |    | 0  |    |
| AB       | 3  | 7  | 0  |    | 0  | 10 |
| MS       | 0  | 0  | 10 |    | 0  | 0  |
| Ce-TZP/A |    |    |    |    |    |    |
| NT       | 2  | 8  | 0  | 10 | 0  | 0  |
| AB       | 1  | 9  | 0  | 0  | 10 | 0  |
| MS       | 0  | 8  | 2  | 8  | 2  | 0  |

A: Adhesive failure at the zirconia-resin interface; C: Cohesive failure within the resin; AC: Combination of adhesive and cohesive failures.
zirconia. As a result of shear testing before thermocycling, the MS group of Y-TZP exhibited significantly higher SBSs than the other groups; however, the MS group of Ce-TZP/A showed lower SBSs than the NT group. After thermocycling, the SBSs of both zirconia specimens, except the MS group of Y-TZP, could not be evaluated, because the resin was already debonded before the tests. Table 8 presents the distribution of the failure modes in the groups. The MS group of Y-TZP tended to exhibit mixed failure (AC) before and after thermocycling, although the other groups of both zirconia types exhibited adhesive failure. SEM observations after the tests revealed that the acrylic resin remained partially in microslits of Y-TZP (Figs. 7 and 8).

**DISCUSSION**

Mechanical interlocking is one of the factors that determine the bond strength between a core material and other materials. For example, improvement of the adhesive strength through the use of retention beads is widely employed for metal-composite restoration. However, many zirconia frames are manufactured using a milling system, and the minimum diameter of milling burs for dental CAD/CAM systems is 0.4–0.6 mm. Therefore, it is impossible to produce slits with widths...
Table 7  Shear bond strengths of acrylic resin veneered to zirconia

| Groups  | 0 Thermocycles | 20,000 Thermocycles | S  | Reduction |
|---------|----------------|---------------------|----|-----------|
|         | Mean | SD  | Category | Mean | SD  | Category |     |           |
| Y-TZP   |      |     |          |      |     |          |     |           |
| NT      | 12.9 | 3.5 | a        | 0    | 0   |          | S   | 100%      |
| AB      | 12.3 | 6.7 | a        | 0    | 0   |          | S   | 100%      |
| MS      | 27.6 | 9.1 | b        | 17.5 | 7.1 |          | S   | 36.6%     |
| Ce-TZP/A|      |     |          |      |     |          |     |           |
| NT      | 16.1 | 2.7 | A        | 0    | 0   |          | S   | 100%      |
| AB      | 11.0 | 4.0 | B        | 0    | 0   |          | S   | 100%      |
| MS      | 8.6  | 5.1 | B        | 0    | 0   |          | S   | 100%      |

SD: Standard deviation; Category: Identical letters indicate that values are not statistically different (p>0.05); S: Significant difference between pre- and post-thermocycling bond strengths (p<0.05).

Table 8  Failure modes of acrylic resin after shear bond testing

| Groups  | Failure mode | 0 Thermocycles | 20,000 Thermocycles |
|---------|--------------|----------------|---------------------|
|         | Failure mode | A   | AC | C | A   | AC | C |
| Y-TZP   |              |     |    |   |     |    |   |
| NT      |              | 10  | 0  | 0 | 10  | 0  | 0 |
| AB      |              | 10  | 0  | 0 | 10  | 0  | 0 |
| MS      |              | 2   | 8  | 0 | 2   | 8  | 0 |
| Ce-TZP/A|              | 10  | 0  | 0 | 10  | 0  | 0 |
| AB      |              | 10  | 0  | 0 | 10  | 0  | 0 |
| MS      |              | 10  | 0  | 0 | 10  | 0  | 0 |

A: Adhesive failure at the zirconia-resin interface; C: Cohesive failure within the resin; AC: Combination of adhesive and cohesive failures.

Fig. 7  SEM images of the debonded surface of Y-TZP veneered with acrylic resin (500×); (A) non-treated, (B) alumina-blasted, and (C) with fabricated microslits before thermocycle, and (D) non-treated, (E) alumina-blasted, and (F) with fabricated microslits after thermocycle.
of just tens of micrometers. Even if microscale milling burs could be fabricated, severe wear of the bur would be expected, because the material considered in this study is fully sintered zirconia. Another approach is laser beam machining, which is commonly used in industry for the manufacture of injection molds, lettering and engravings, and can also produce microscopic surface texture on injection molds. This eliminates the need for burs. Therefore, this technology has the potential to make microslits on zirconia frames having complex shapes. For these reasons, laser beam machining was selected as the fabrication method for microretention features in this study. Microslits were fabricated on zirconia disks with flat surfaces and the SBSs of porcelain, indirect composite resin and acrylic resin veneered to the zirconia disks were measured. The microslits on the zirconia disks were fabricated in a simple reticular pattern because the purpose was to examine the SBS with or without microslits.

In general, since resins exhibit a relatively high level of water absorption, and their coefficient of thermal expansion (CTE) is higher than that of zirconia, a reduction in adhesion strength might be expected as a result of long-term exposure to moisture and significant temperature changes in the oral cavity. Therefore, the change in the SBS before and after thermal cycling was examined for indirect composite resin and acrylic resin. Because porcelain exhibits relatively low water absorption and its CTE is close to that of zirconia, no thermal cycling was performed on porcelain samples.

Effects of microslits on porcelain veneered to zirconia

We previously hypothesized that the strength of the bond between zirconia and porcelain would improve if the zirconia surface roughness was increased, and investigated the effect of alumina blasting with particles of different sizes on the SBS. However, there was no significant difference between the different particle sizes. Thus, the present study was carried out to investigate whether the bond strength could be improved by introducing a more geometric surface texture. After shear tests, there was no difference evident between the groups. However, in the MS group, SEM observations of both types of zirconia after the shear tests indicated that the microslits were still totally covered with the porcelain, i.e., the porcelain remaining on the zirconia may have been due to the microslits. Accordingly, this result indicated the microslits have the effect to hold the porcelain, although the SBS values were not increased by the microslits. Henriques et al. reported that Nd:YAG laser surface structuring of zirconia increased the SBS with porcelain by up to 75%. They produced holes in the zirconia surface with diameters of 25 and 50 µm, and depths of about 50 and 100 µm, respectively. Although the porcelain used in the study was veneered using an injection technique, which is different to our method, deeper microslits may be more effective for increasing the SBS.

The reason why there was no difference from NT despite the fact that all MS specimens underwent cohesive failure was that the mechanical strength of the porcelain penetrating the microslits was weaker than the adhesive strength. As a result, the strength of the porcelain was close to the SBS for the NT specimens. This may be due to the brittleness of the porcelain or the design of the microslits. Therefore, it may be necessary to strengthen the porcelain, or modify the microslits, for example by making them wider or changing their shape.

Effects of microslits on indirect composite resin bonding to zirconia

Previous studies have suggested that 10-methacryloyloxydecyl dihydrogen phosphate (MDP)
bonds chemically to zirconia. MDP consisting of phosphate monomers is effective to improve the bond strength between indirect composite resin and zirconia. Therefore, in this study, a primer including MDP was used as a chemical treatment to bond indirect composite resin to zirconia.

The SBSs of both types of zirconia of the NT group were significantly reduced after thermocycle, even though the SBSs before thermocycle were over 15.0 MPa. In contrast, both types of zirconia of the AB and MS groups exhibited measurable SBSs after thermocycle. This result demonstrates that the requirement for improvement of the bond durability is not only chemical treatment, but also surface roughening such as alumina-blasting and the fabrication of microslits.

Furthermore, the SBS of Y-TZP of the MS group was significantly higher than that of the AB group, regardless of thermocycling. However, Ce-TZP/A of the AB group before and after thermocycling showed the highest SBS. In addition, all specimens of the MS group with Y-TZP showed cohesive failure, whereas specimens on Ce-TZP/A showed adhesive and combined failure. Thus, the effectiveness of the microslits on Y-TZP could be greater than those on Ce-TZP/A. This is attributed to the difference in the structure between two types of zirconia. Figure 9 shows SEM images of texture inside microslit. According to the SEM images, the microslits on Y-TZP were relatively rougher than those on Ce-TZP/A. This difference might be due to difference of the crystal structure between two types of zirconia. Xie et al. reported that hot acid etching increased the surface roughness of Y-TZP, and improved the bonding with resin, which suggests that the surface roughness plays a significant role in the bonding. Therefore, the present result suggested that not only the design of the microstructure but also the texture inside the slit may affect the bond strength, and further investigation is needed.

Effects of microslits on acrylic resin bonding to zirconia

In the present study, a primer containing MDP was applied to examine the effect of the microslits on the SBS between zirconia and acrylic resin. Several studies have reported that a primer that contain MDP is recommended for bonding zirconia with acrylic resin. Accordingly, an alloy primer that contained MDP was used for this experiment, as with the third experiment.

The relationship between load and the flexibility of both Y-TZP and Co-Cr has been previously reported using a four-point bending test. The study reported that Y-TZP showed an equal maximum load but less flexibility than Co-Cr. However, the deformation of zirconia when it is packed in a flask and pressure is applied has not been clarified. Therefore, this study investigated the application of acrylic resin as a cast polymerization.

After thermocycles were applied to Y-TZP, the SBS of the specimens of the NT and AB groups could not be evaluated, because the resins were already debonded prior to the tests. In contrast, the MS group retained the resin on the Y-TZP surface after thermocycle, and the MS group exhibited a measurable SBS. In addition, under SEM observation, residual resin was found on some parts of the fracture surface of the MS group, before and after thermocycles. These results indicated that the microslits are superior for bond durability between Y-TZP and acrylic resin. Kawaguchi et al. reported on the bonding of acrylic resin to Ti and Co-Cr alloy with four surface treatments. The result of their studies indicated that the use of a primer containing MDP yielded the highest bond strength of acrylic resin to both Ti and Co-Cr alloy after thermocycling. The bond strength of Ti is similar to that of the MS group of Y-TZP examined in the present study. Therefore, there is the possibility that the microslits on Y-TZP are effective for bonding acrylic resin to Y-TZP, as with the bond stability of removable partial dentures fabricated with acrylic resin and Ti.

However, among the Ce-TZP/A specimens, the SBS of the MS group before thermocycle was the lowest, followed by the AB group. Before shear tests, the specimens of the MS group were fractured by thermocycle stress, as with the other groups. In addition, SEM observations after shear tests showed that no residual resin was remained on the MS group zirconia surface. This is probably because, similar to the case for the indirect composite resin, the microslits on the Ce-TZP/A were smoother than those on the Y-TZP (Fig. 9). The surface texture inside the microslits may be an important factor in the case of an acrylic resin because the SBS after thermal cycling decreased significantly.

In this study, laser machining was performed in an area with a diameter of 5 mm, and the processing time was about 5 min. Since the machine used is for industrial applications, it is not yet easy to modify it for clinical use. It is necessary to improve the machining time and cost. However, we successfully demonstrated that a Nd:YVO4 laser can engrave microscale structures in fully sintered zirconia. In particular, microslits on Y-TZP were shown to significantly improve the adhesion to indirect composite resin and acrylic resin. In the future, smaller machines are likely to be developed that are easier to use for dentistry, and can operate at lower cost.

Although the microslits had no effect on the bond strength with porcelain, they were particularly effective for Y-TZP for the two types of resin. This may be due to the difference in toughness between porcelain and
resin. Because porcelain is brittle, the material that penetrates into the slits may be easy to break. Therefore, the slit shape may be inappropriate for porcelain. On the other hand, probably because the indirect composite resin and acrylic resin have a relatively high elongation at break, the microslits improved the bond strength. Although microslits were effective under some conditions, the SBs may be further improved by modifying their microstructure, for example by making them wider and deeper. In addition, from the results for indirect composite resin and acrylic resin veneered to Ce-TZP/A, the texture inside the microslits may also be an important factor for improving the bond durability. Further investigation is needed to compare the influence of the microslit shape on the bond strength.

CONCLUSIONS

Within the limitations of the present study, the bond strength between porcelain and two types of zirconia was not improved by the use of microslits. In addition, the microslits on Ce-TZP/A did not improve the bond strength with acrylic resin, despite showing a better effect than non-treatment for adhesion to indirect composite resin. However, microslits on Y-TZP were found to improve the bond strength and adhesion durability in bonding with indirect composite resin or acrylic resin. The slit-like microstructure on zirconia partially improved the adhesive strength and durability. However, it is necessary to further investigate the effect of different microstructures on zirconia, for example wider and deeper slits or other shapes, on the bond strength between zirconia and porcelain or indirect composite resin or acrylic resin.

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